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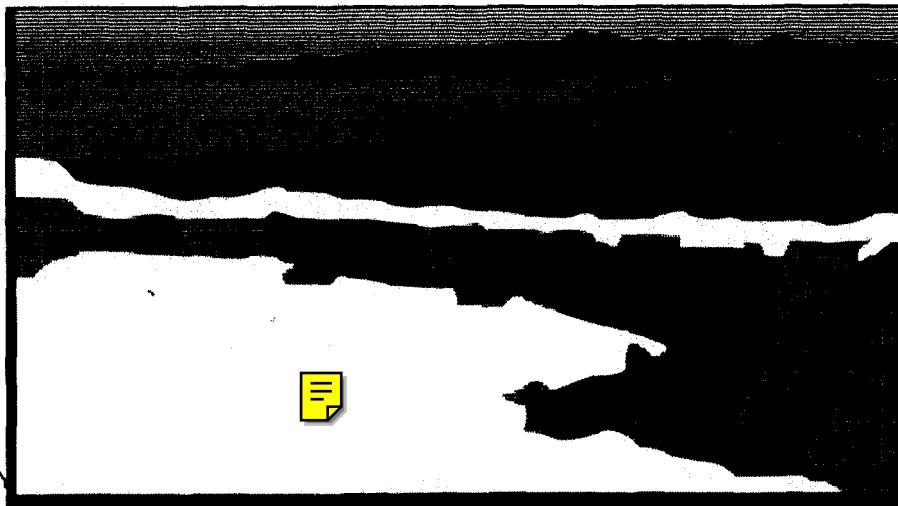
Title: A THEORETICAL EQUATION OF STATE FOR
DETONATION PRODUCTS WITH CHEMICAL
EQUILIBRIUM COMPOSITION OF THE SURFACE OF
SMALL CARBON CLUSTERS

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A THEORETICAL EQUATION OF STATE FOR DETONATION PRODUCTS WITH CHEMICAL EQUILIBRIUM COMPOSITION OF THE SURFACE OF SMALL CARBON CLUSTERS *

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Abstract. A theoretical equation of state for detonation products is described and compared with various data. A perturbation theory approach is used for the mixture of molecular fluids that is based on Monte Carlo simulations. The solid carbon is characterized as small clusters rather than as a bulk material. The free energy of the clusters is modeled as primarily additive contributions such as vibrational modes, bond strengths, and effective volumes. Up to 30% of the atoms in a diamond cluster are on the surface with dangling bonds capped by various groups composed of C, H, N, and O from the background fluid. A counting term similar to ideal entropy of mixing is also found for the surface composition. Competition between the U, TS, and PV terms in the Gibbs free energy leads to dramatic shifts in the surface composition in some regions. This in turn leads to shifts in the background fluid mixture composition and anomalous behavior in the total EOS. This behavior in PBX 9502 is in good agreement with recent data on release isentropes from overdriven states(1).

INTRODUCTION

The detonation products of a high explosive form a very nonideal and complicated mixture under extreme conditions ($P \approx 30$ GPa and $T \approx 3000$ K). An accurate theoretical treatment must not only handle well the extreme conditions for a single molecular fluid, but also the mixture of species and the thermodynamic chemical equilibrium composition of the products. In addition, the solid carbon is in the form of small clusters with a large fraction of atoms on the surface. In the diamond phase, the spherical clusters range from 20 - 50 Å in diameter with 20 - 30 % of the atoms on the surface with energetically unfavorable dangling bonds.

In this work, we provide for a variety of dangling bond cap groups in chemical equilibrium. The background fluid treatment is given by a fluid perturbation theory tied to benchmark simulation methods.

CARBON CLUSTERS

In this section we focus on the treatment of diamond clusters. A more detailed description can be

found in a separate paper(2). (Graphite has no dangling bonds except at the edges and is treated as a bulk graphite EOS with a small shift in the heat of formation to include restructuring at the edges.) A spherical diamond cluster has dangling bonds exposed over the entire surface. It is energetically unfavorable to restructure the diamond surface(3). Figure 1 shows a slice through the $z=0.5$ (lattice constant) plane for a 23 Å cluster. A mirror plane is indicated as a dashed line. Surface atoms with single and double dangling bonds (denoted 1 and 2) are shown in the upper left. Examples of possible surface groups are shown in the corresponding mirror sites in the lower right. Nonsurface atoms, denoted 'core carbon atoms', are indicated by a circle and the origin is given by a plus. Also note that the tetrahedral bonds are all out of the plane and none of the atoms in a plane are bonded to each other.

For a given radius cluster, the number and type of surface groups in the truncated lattice and the number of core atoms without dangling bonds are counted. Figure 2 shows the ratio of surface to core atoms as a function of the number of core atoms for discrete possible clusters. The line shows an analytic fit that is

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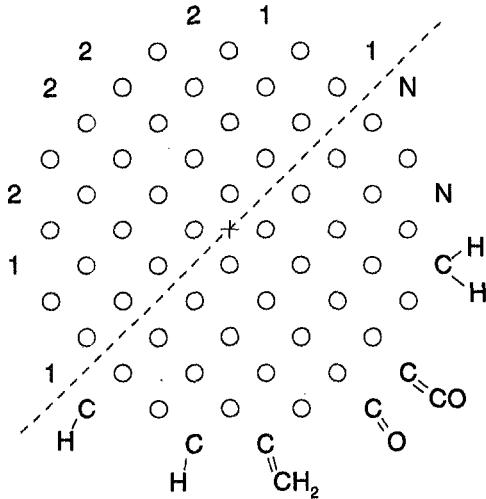


FIGURE 1. The $z=0.5$ plane of a 23 \AA cluster. A mirror plane is indicated as a dashed line. Surface atoms with single (1) and double (2) dangling bonds in the upper left. Examples of surface groups in the corresponding mirror sites in the lower right.

dominated by the geometric surface to volume ratio of a sphere. The relative fraction of single dangling bond sites is nearly constant at about 0.65 over a wide range of cluster size.

In order to determine the equilibrium composition of the surface, it is necessary to determine the Gibbs free energy of the cluster with arbitrary surface composition. Figure 3 shows a diagram of a small part of the surface. Each surface group has an additive contribution to the free energy. An Einstein model is used for the vibrational frequencies in surface group. These modes are assumed to couple weakly to the core and are estimated from similar small molecules where possible. The heat of formation of each surface group is determined by additive bond strengths. The effective volume is estimated from the geometry of similar groups in small molecules. An effective standoff distance from the geometric surface is chosen to take into account that the occupied volume (i.e. volume excluded from the molecular fluid) is not identical to the geometric volume. Notice that a

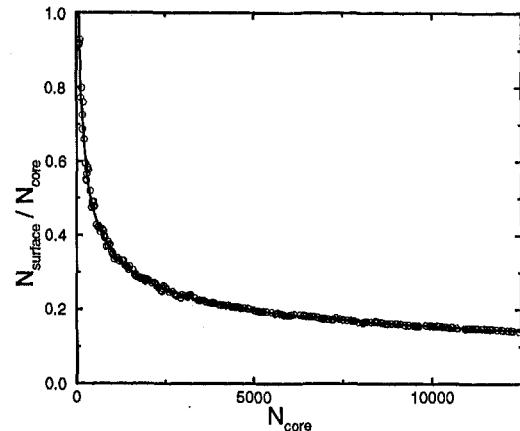


FIGURE 2. Ratio of surface sites to core atoms versus the number of core atoms in a cluster

substitutional N atom would occupy a small volume that will become significant in the PV term at high pressure.

The surface free energy has an entropy contributions due to the counting of possible ways to occupy a set of surface sites with a given composition. This contribution to the Helmholtz free energy for single bond sites is given by

$$\frac{A^{(1)}}{NkT} = \sum' x_i^{(1)} \ln(x_i^{(1)} / x^{(1)}) \quad (4)$$

where $x_i^{(1)}$ is the concentration (mol) of surface groups of type i and $x^{(1)}$ is the total concentration of single dangling bond sites (and similarly for the double bond sites).

Finally, the core contributions are taken from a bulk diamond model incorporating a Debye model of vibrations and a measured cold curve. A related model by van Thiel and Ree(4) implemented similar characterization of the surface groups with a fixed choice of surface composition.

FLUID MIXTURE

Benchmark simulation methods are used to test the various approximations used to evaluate the fluid mixture EOS. Accurate fluid perturbation theories

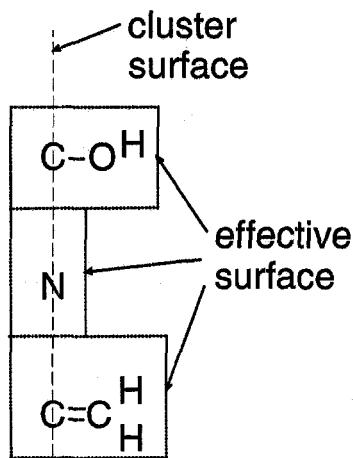


FIGURE 3. Diagram of the effective surface and occupied volume (box) of additive surface groups

and other approximations are used for spherical potentials, nonspherical potentials, nonideal mixing, and chemical equilibrium composition. Details are given elsewhere(2) because of space limitation.

The parameters in this method were least squares fit over a limited set of data and with terms added to limit the variation of the constants where an estimate could be made of the value and uncertainty. The slope change in the overdriven Hugoniot of PBX 9502 had a strong influence(2) on the final set of parameters. Further refinement is planned with a much larger data set, in particular using more of the new types of measurements such as sound speeds(5) and temperatures(6).

RESULTS

Vorthman et al.(1) have measured the release isentrope in PBX 9502 from overdriven states. The anomalous behavior of the data at high pressure is shown in Fig. 4. The present theory is in very good agreement with the data and provides a possible mechanism for the anomalous behavior. At high pressure, the single dangling bond surface C atoms are replaced by a substitutional N atom with no dangling bonds. The PV term gives preference for the small volume occupied by this surface group. At low pressure, the lower energy of a H cap on the dangling

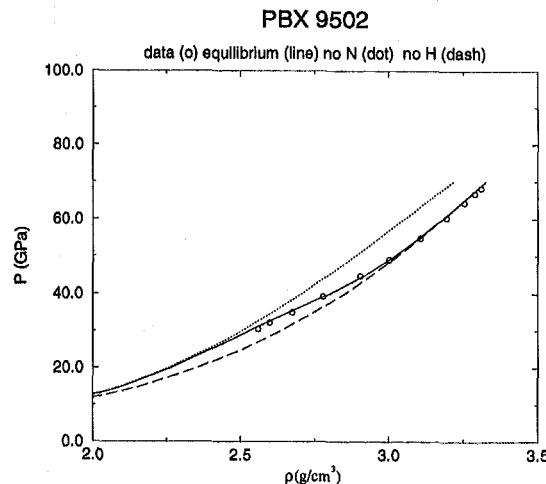


FIGURE 4. PBX 9502 isentrope from a high pressure overdriven state. Data (O), Theory: equilibrium (line), substitutional N (dash), H cap (dot).

bond dominates. The transition in surface chemistry is spread over 20 GPa in pressure and 0.5 g/cm^3 in density. The theory reproduces the rapid slope change around 50 GPa. In the transition region, the slope of the isentrope is very nearly constant. That implies the sound speed is also nearly constant over that region. Figure 5 shows the rapid variation in surface composition that produces the unusual EOS.

There is also a very large effect due to the graphite diamond transition. Figure 6 shows good agreement between theory and experiment(7) for a series of detonation velocity versus density for TNT, 25/75 TNT/RDX, 50/50 TNT/RDX, and RDX. The detonation velocities are dominated by the graphite diamond transition. Although the transition region is rather small (1 - 2 GPa), Fig. 7 shows that the minimum shock velocity is sometimes determined by the location of the equilibrium transition rather than the minimum of either phase.

CONCLUSIONS

The chemical equilibrium treatment of surface chemistry on diamond clusters was started based on the premise that there was too large a fraction of atoms on the surface to treat the carbon as a pure bulk material. The calculations demonstrate that there can

PBX 9502

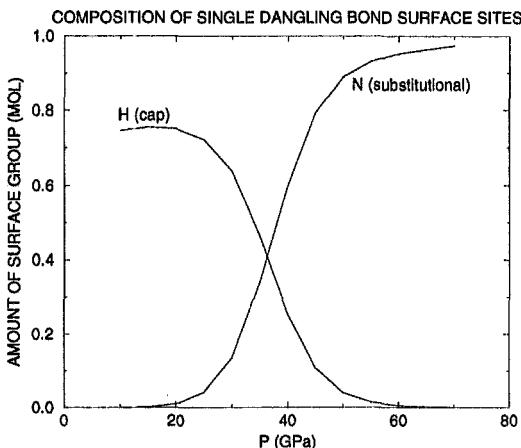


FIGURE 5. Composition of single dangling bond surface sites. Moles of surface group per formula weight versus pressure on the isentrope shown in Fig. 6.

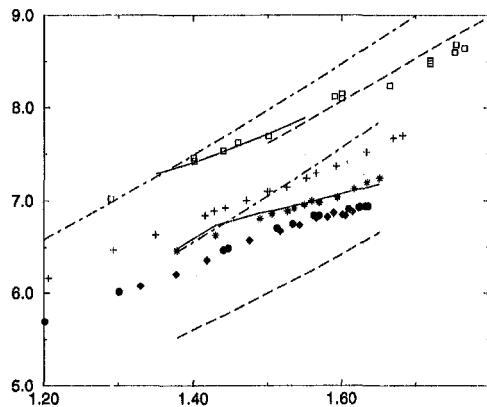


FIGURE 6. Detonation velocity versus density for PBX 9502. Theory: equilibrium (line), graphite only (chain dash), and diamond clusters only (dash).

be regions with a large shift in the chemical composition on the surface and that the consequences of these shifts are within the range of sensitivity of ex-

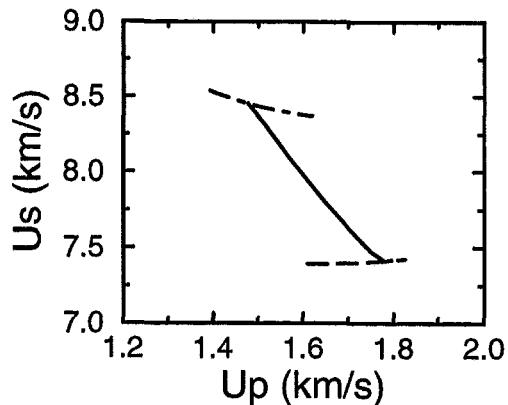


FIGURE 7. Shock velocity versus particle velocity at two values of initial density of RDX in the transition region. Equilibrium (line), graphite only (chain dash), and diamond only (dash). CJ state for equilibrium (O), graphite only (+), and diamond only (*)

periments. These preliminary results suggest that fitting an EOS theory to detonation velocity data is not as straightforward as it seems. Significant portions of the data could be strongly influenced by transition regions in the chemical composition.

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**1999 TOPICAL CONFERENCE
ON SHOCK COMPRESSION OF CONDENSED MATTER**

June 28 - July 2 1999

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OVERVIEW

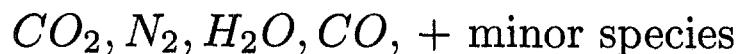
DETONATION PRODUCTS

- Unreacted:



- Products:

dense fluid mixture



carbon clusters ≈ 1000 atoms $\approx 20 \text{ \AA}$

graphite-like

diamond-like

APPROACH

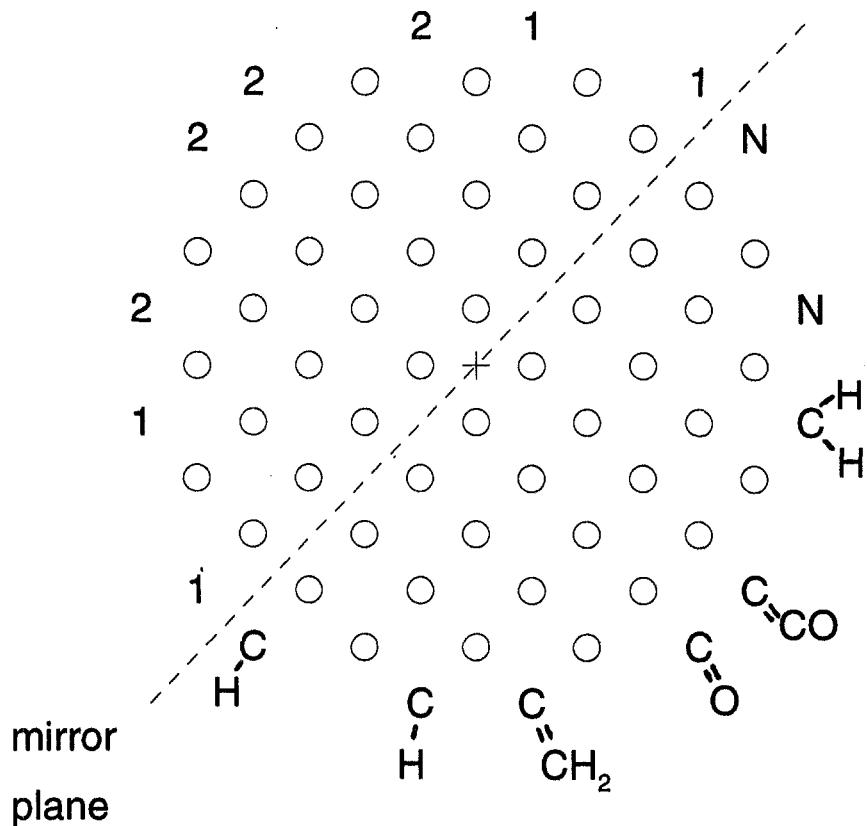
- Characterize Cluster Surface
- Chemical Equilibrium of Surface
- Simulation Methods
- Perturbation Theory
- Constrained Fit

CROSS SECTION OF A SMALL DIAMOND CLUSTER

CORE ATOMS: 0 = 0 DANGLING BONDS

SURFACE ATOMS: 1 = 1 DANGLING BOND

2 = 2 DANGLING BONDS

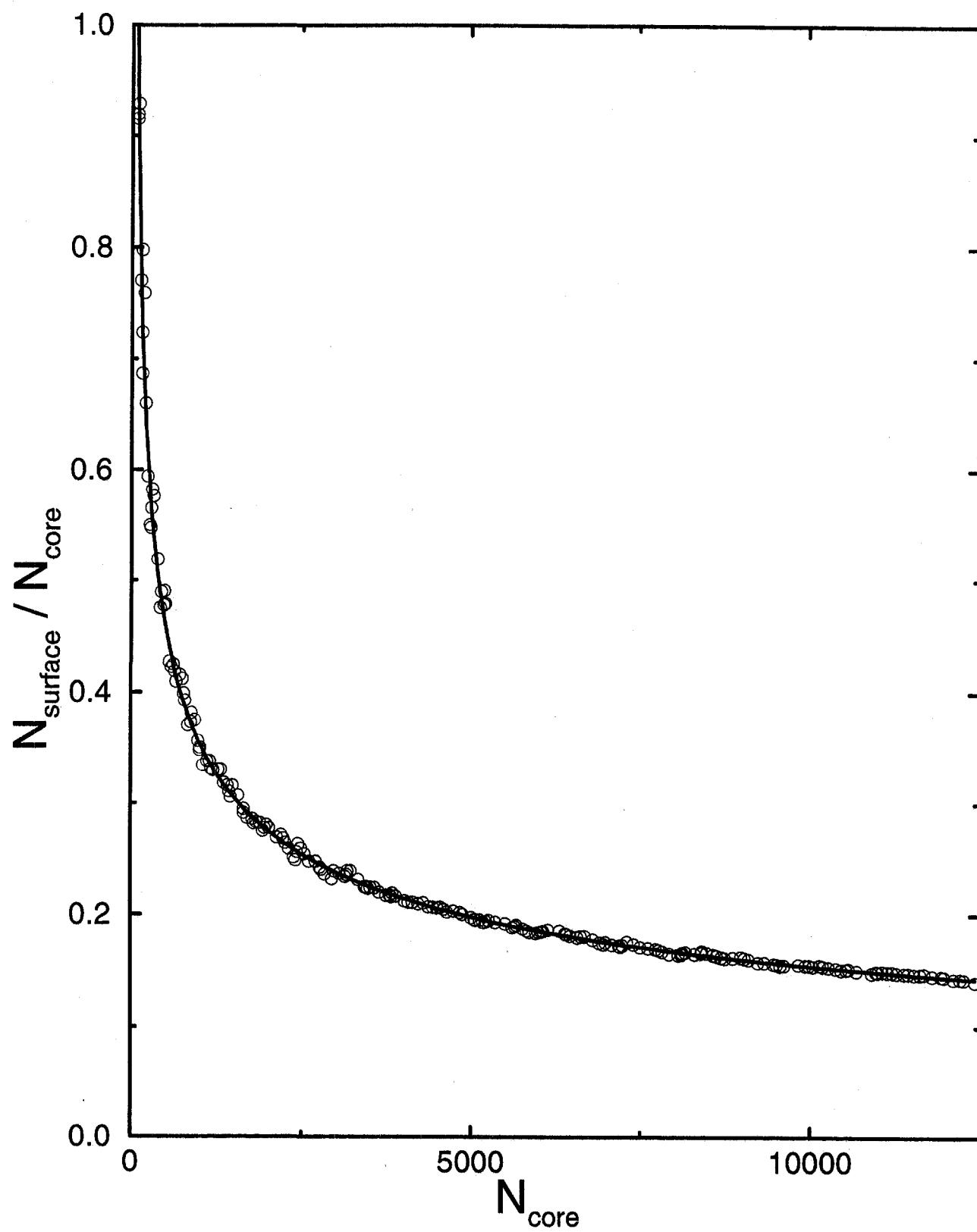


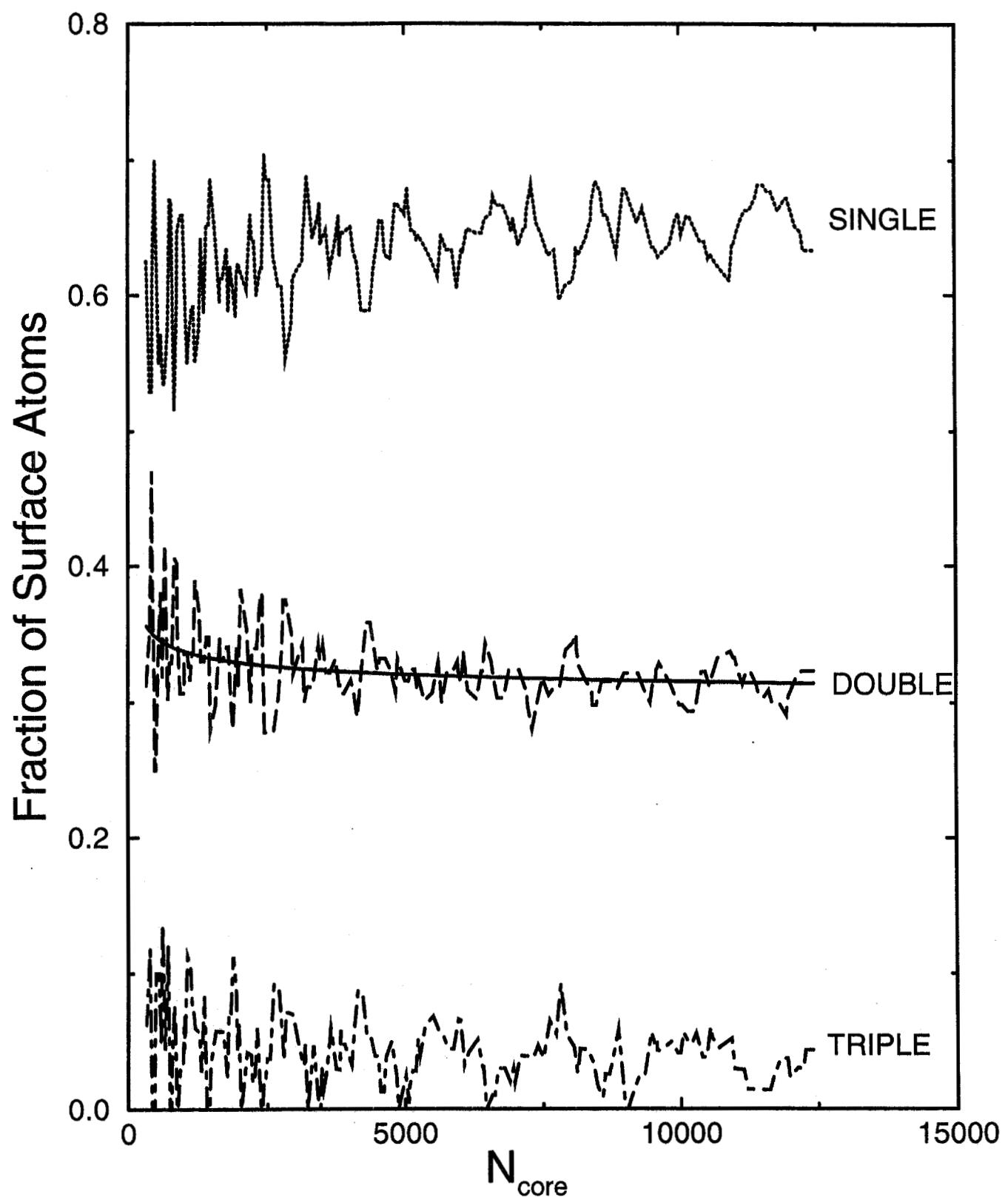
EXAMPLES OF SURFACE GROUPS

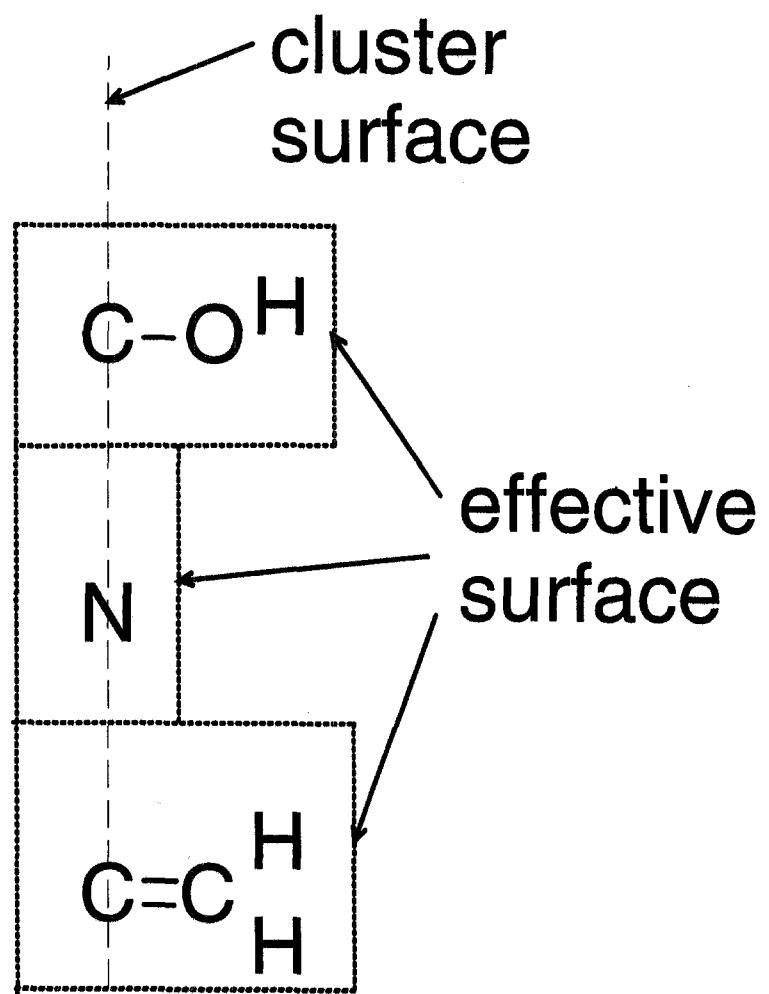
note: N without dangling bonds

can substitute for a surface C

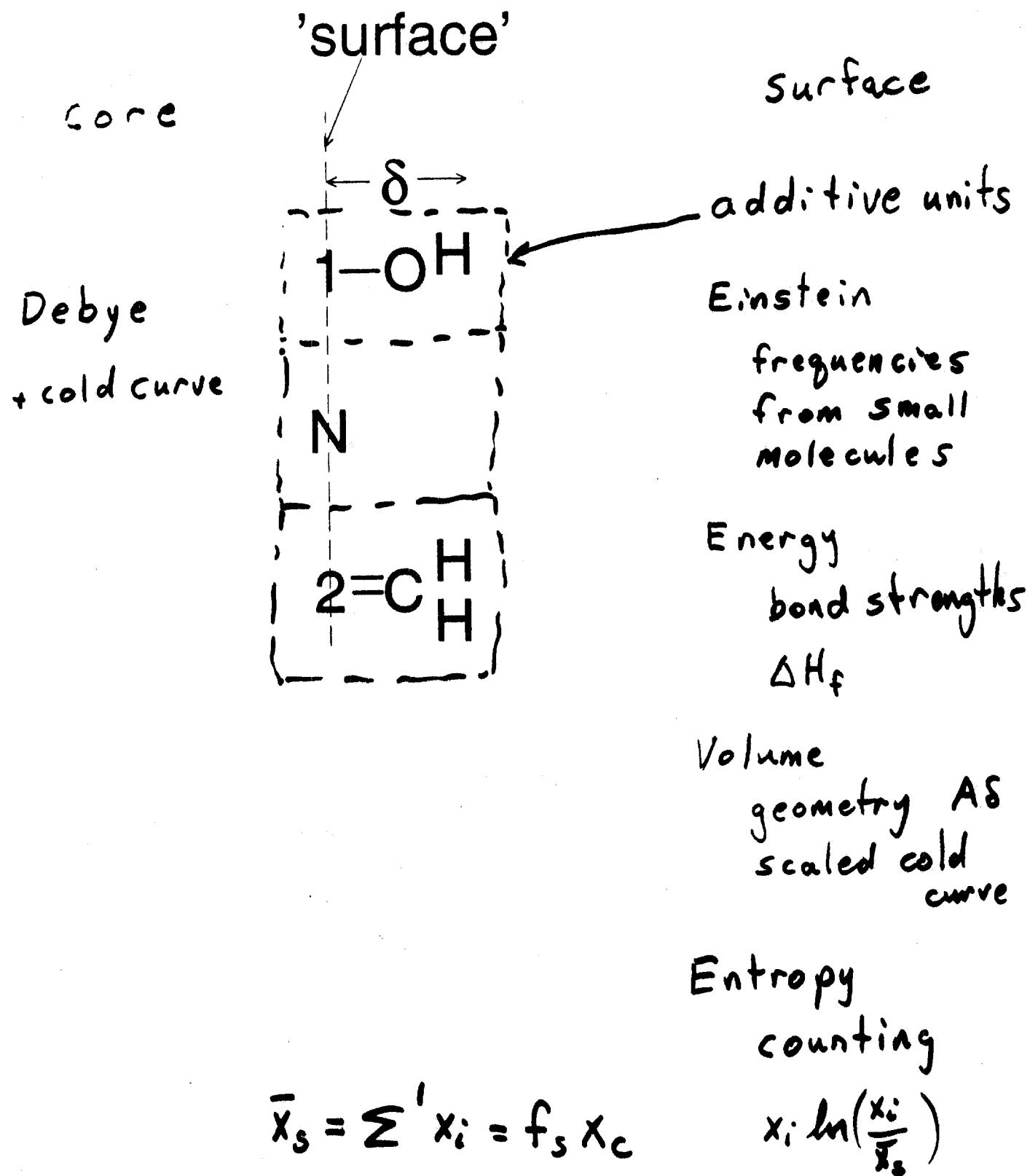
with one dangling bond





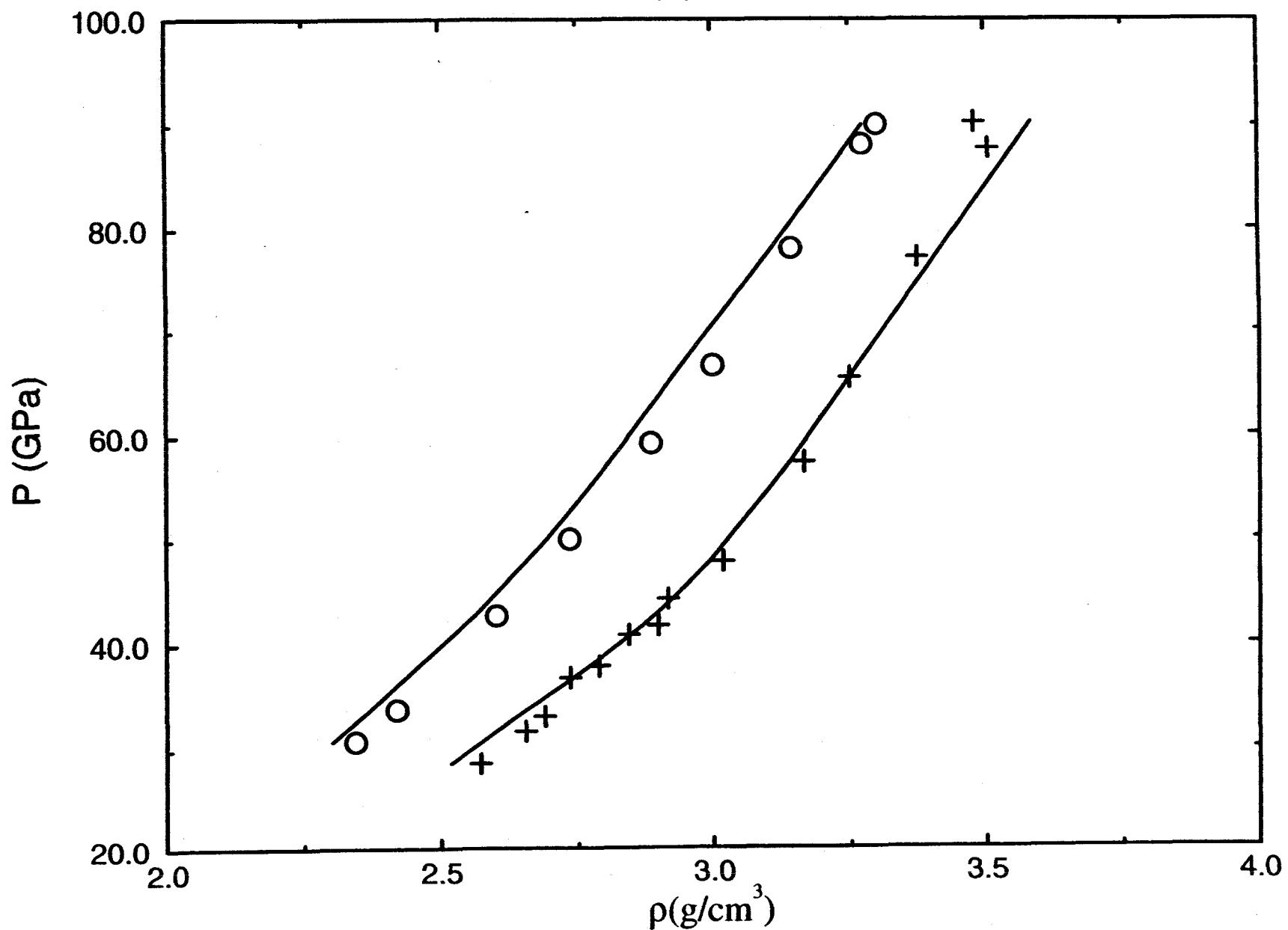


$$G(x_1, \dots, x_n, P, T)$$



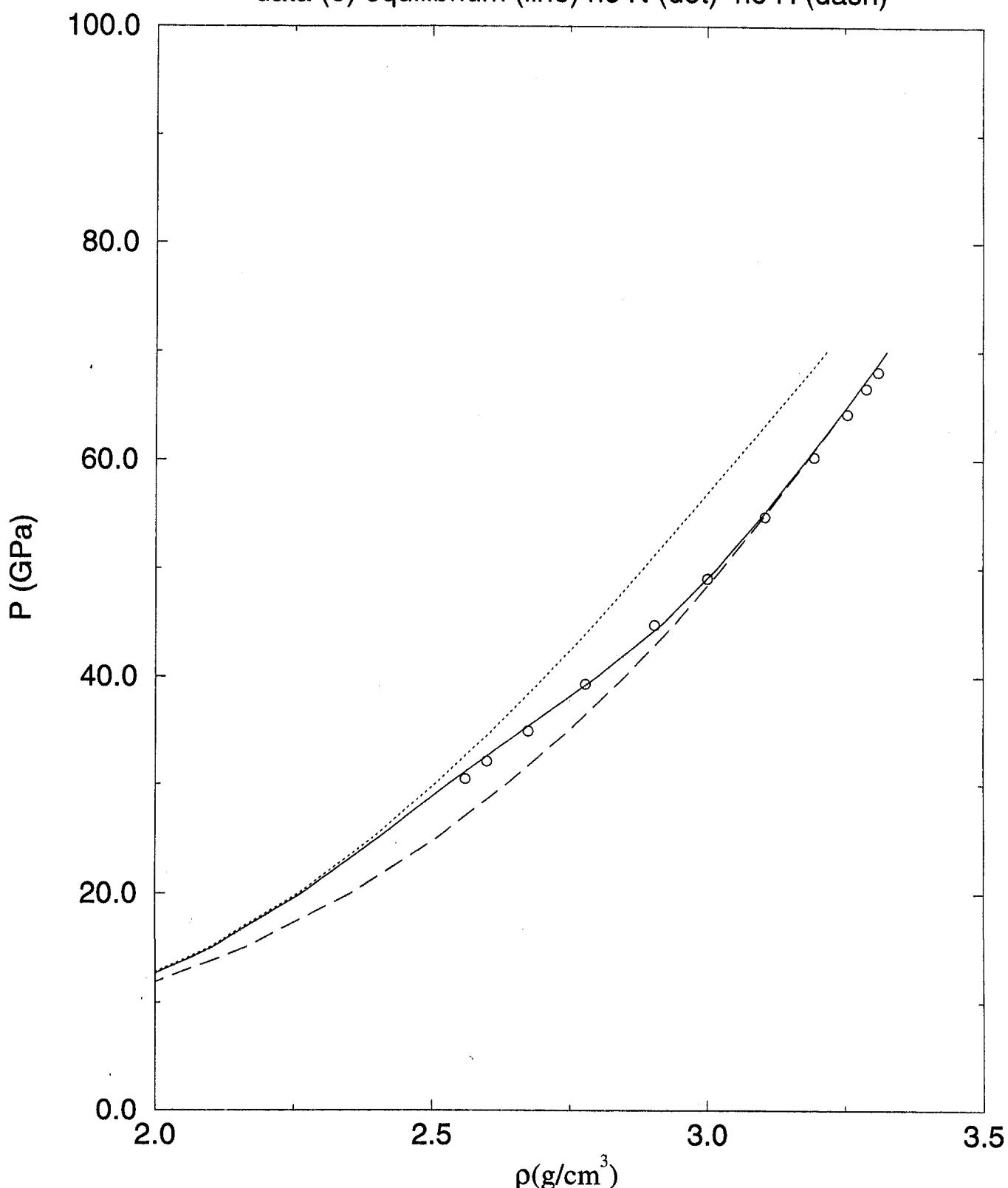
OVERDRIVEN HUGONIOT

PBX-9501 (O) PBX-9502 (+)



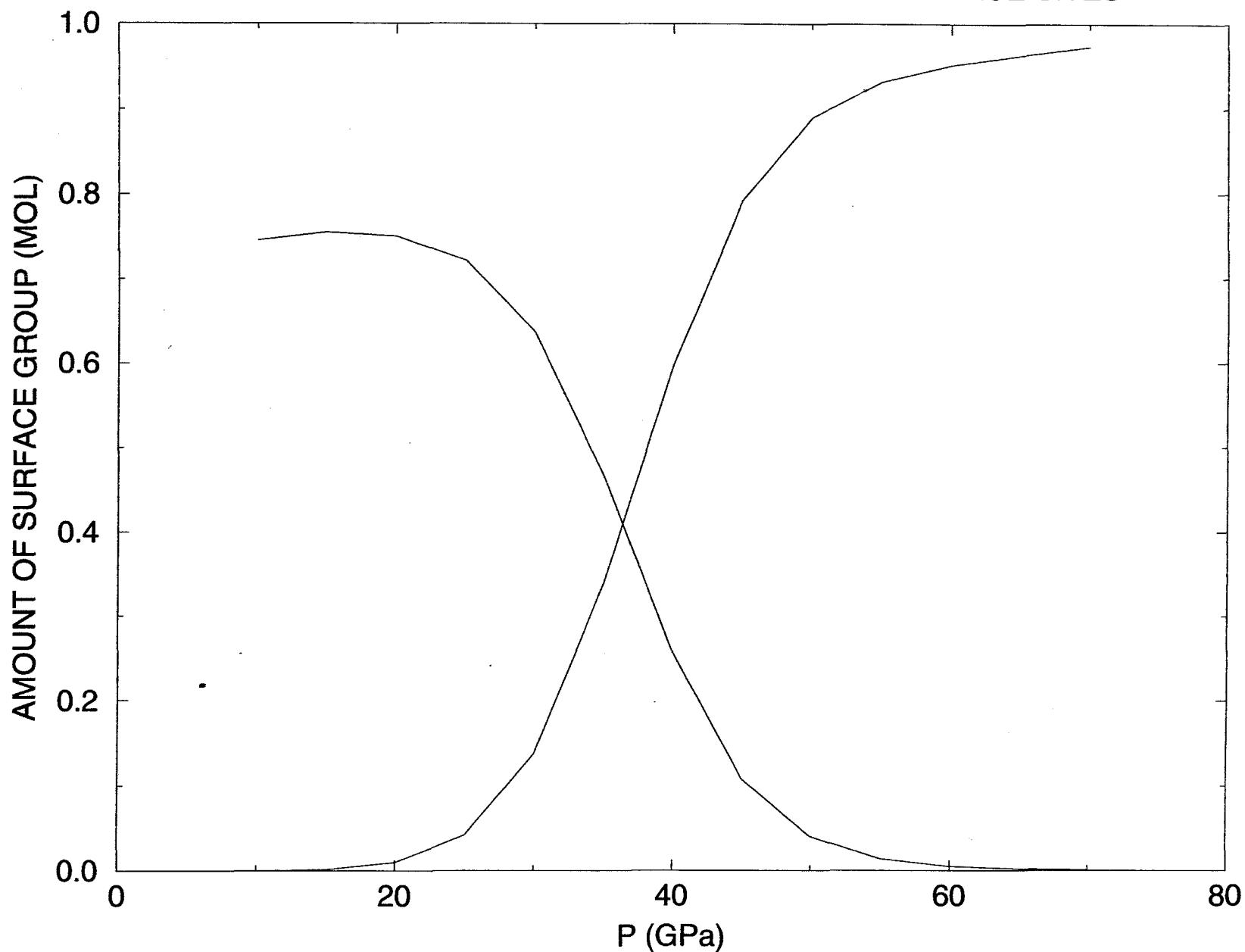
PBX 9502

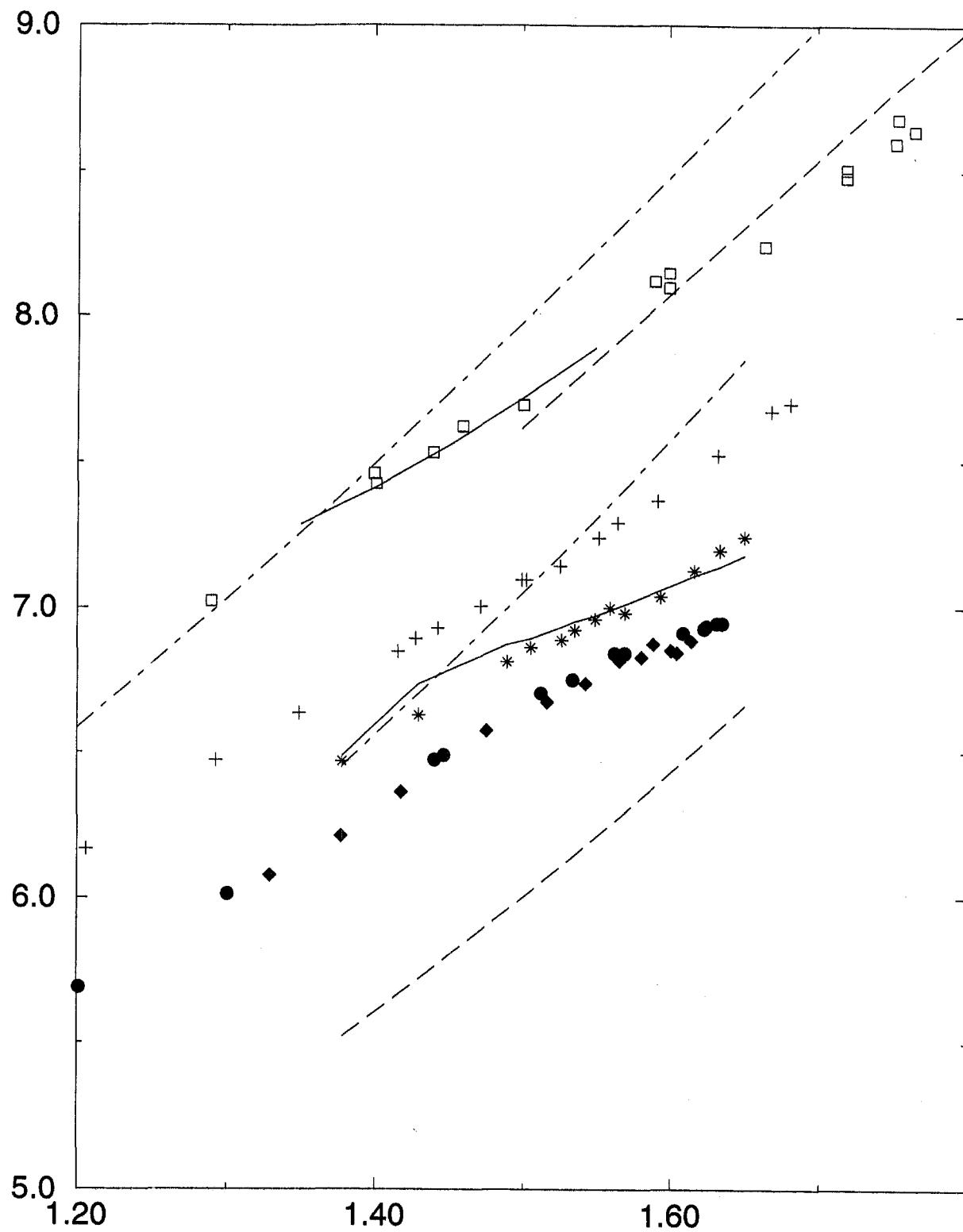
data (o) equilibrium (line) no N (dot) no H (dash)



PBX 9502

COMPOSITION OF SINGLE DANGLING BOND SURFACE SITES





50/50 RDX/TNT

Equilibrium (line) Diamond (dash) Graphite (chain)

