

Swelling Behavior of Halthane 73-18 Polyurethane Adhesive in Dimethyl Sulfoxide (DMSO)

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ADHESIVE IN DIMETHYL SULFOXIDE (DMSO)**

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Introduction

To insure safe performance during the launch and flight of the W79 Artillery Fired Atomic Projectile (AFAP), the assembly gaps in the high explosive assembly were filled with a continuous film of polyurethane elastomer adhesive called Halthane 73-18. To disassemble bonded weapons like the W79, Lawrence Livermore and Mason & Hanger, Pantex Plant have developed a chemical dissolution process that safely removes the high explosive, thereby facilitating the recovery of the pit. The solvent of choice for the W79 AFAP was dimethyl sulfoxide (DMSO).

In the W79 dissolution process, a continuous spray of DMSO is emitted through nozzles mounted in manifold assembly that encircles the HE assembly. The operating pressure and temperature of the DMSO are less than 100 psig and less than 160°F. Although warm DMSO readily dissolves the LX-10¹ explosive, it cannot dissolve the Halthane 73-18 adhesive due to its chemically crosslinked structure. DMSO does, however, swell the Halthane adhesive. The resulting swollen films are soft and unable to support their own weight, yet they are not necessarily so fragile that they will tear or shred readily under the force of the DMSO spray. Indeed, the swollen Halthane films encountered in several W79 Type 6B units tested in the Pantex Workstation proved to be quite tenacious. They remained intact under the action of DMSO spray and became an encapsulating barrier that shielded the remaining undissolved HE. This effectively stopped the dissolution process, forcing manual removal in order to complete the dissolution process. By comparison, the swollen Halthane film was readily shredded and eliminated under the action of the DMSO spray nozzles in tests at LLNL in workstation of a different design. This apparent difference in response is the subject of this report.

Background

At the time that the work described in this report was being performed, two dissolution workstation designs were being evaluated for possible use in the W79 AFAP dismantlement program, one at Pantex herein called the *Pantex Workstation*, and the other at Site 300 at Lawrence Livermore National Laboratory herein called the *LLNL Workstation*. Since this time, the LLNL-designed workstation was selected for the W79 Dismantlement Program and it is currently being installed at Pantex. Although the final version of the LLNL Workstation contains some significant design changes from the development version used in this study, the general dissolution process and the DMSO spray conditions remain the same.

At the time of the tests described in this report (Spring/Summer 1994) there were several potentially significant differences between the Pantex and LLNL Workstations. These included 1) the DMSO pump

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¹ LX-10 is 95 wt% HMX, 5 wt% Viton-A

pressure, 2) the orientation of the spray nozzles, 3) the spray nozzle patterns, and 4) the origin of the test W79 assemblies. More details of these differences are now discussed.

The LLNL Workstation employed a higher pump pressure than Pantex Workstation, about 65 psig versus ~35 psig. Pantex apparently opted to employ a lower pressure out of concern for the longevity of their pump. Combined with a higher flow rate and an optimized nozzle pattern employed in the spray manifold, it appeared that the DMSO in the LLNL system was capable of applying more force and thereby doing more work on the swollen Halothane film (i.e., tearing and shredding).

In the Pantex Workstation, the orientation and low delivery pressure of the spray nozzles permitted the Halothane film to drape over the assembly and shield the undissolved HE from the DMSO spray. This effectively stopped the dissolution process. In contrast, the LLNL Workstation used a nozzle orientation and delivery pressure that facilitated the dissolution process by shredding the Halothane and allowing DMSO to rapidly drain away from the assembly, thus continuously exposing the undissolved HE to the DMSO spray.

One last possible difference -- the one addressed in this study -- was that there was some difference in the Halothane 73-18 adhesives in the W79 units tested in the LLNL Workstation versus those tested in the Pantex Workstation. In development testing, LLNL used freshly built non-nuclear assemblies constructed at Site 300, whereas the Type 6B W79 units used by Pantex were "naturally aged" flight test units that were built at the same time as the W79 warheads. Because polymers can undergo chemical changes during aging, such as degradation and/or crosslinking, it was deemed a possibility that the Halothane 73-18 in the LLNL assemblies might not be representative of the materials in older Type 6B units. For this reason the effects of thermally accelerated and natural aging on the swelling behavior of Halothane 73-18 were examined in this study.

Experimental

Specimens

Five types of Halothane 73-18 specimens were evaluated in this investigation.

- *DMSO Swollen Films* were recovered from W79 assemblies tested at Pantex and LLNL.
- *Retention Samples* were recovered from kit mixes used to assemble LLNL test units at Site 300.
- *Lab-Prepared Halothane* samples were at LLNL in the form of molded sheets/films 0.005- 0.020 in thick, and bulk slabs 0.16 in thick.
- A slab sample of seventeen-year-old "*Naturally Aged*" Halothane was obtained from Y12.² These were prepared at room temperature (R.T.) in 1977 and thereafter stored in the lab under ambient conditions.
- "*Used*" Films (~0.005 in thick) were recovered from Halothane bonded lap shear specimens that were prepared and tested at LLNL.

Swelling Properties

Simple swelling and drying measurements were carried out on dry and swollen Halothane 73-18 specimens using pure DMSO. These specimens typically were between 0.2-1 gram in mass. Equations 1-3 describe the properties measured. The "sol fraction" described by Equations 3 and 4 is that portion of the Halothane that is extracted from the sample by the DMSO. The sol fraction is a function of the weight gain on swelling and weight loss on drying as shown in Equation 4. Note that the sol fraction and

² Provided by George Dorsey at the Y12 Plant in Oak Ridge.

weight gain on swelling and weight loss on drying as shown in Equation 4. Note that the sol fraction and weight gain cannot be measured for swollen films recovered from the Pantex and LLNL Workstations because the initial dry weight, W_o , is unknown

Measured Swelling Properties				
• <i>Weight Gain on Swelling</i>	Gain (wt%)	$= 100 \times (W_w - W_o) / W_o$	Positive value	Eq 1
• <i>Weight Loss on Drying</i>	Loss (wt%)	$= 100 \times (W_d - W_w) / W_w$	Negative value	Eq 2
• <i>Sol Fraction</i>	Sol (wt%)	$= 100 \times (W_d - W_o) / W_o$	Negative value	Eq 3
		$= 100 \times ((1 + \text{Gain}/100)(1 - \text{Loss}/100) - 1)$		Eq 4

W_w = Equilibrium swollen weight, W_o = Initial (dry) weight, W_d = Dried weight

To obtain the equilibrium swollen weight, W_w , the following procedure was used: 1) Measure initial (dry) weight, W_o . 2) Immerse in neat DMSO in a tightly sealed glass vial for 24-48 hours at 50°C (elevated temperature used to accelerate equilibrium swelling). 3) Remove vial from oven and allow to cool to room temperature. 4) Remove specimen from DMSO with forceps and pat dry on lintless paper towel to remove excess liquid DMSO. 5) Weigh to get W_w .

To obtain the weight loss on drying, W_d , the following procedure was used: 1) Place swollen specimen on a piece of pre-weighed aluminum foil. 2) Dry specimen in 50°C forced-air oven for 16-24 hours. 3) Transfer specimen to 50°C vacuum oven for final drying. 4) Weigh specimen periodically to constant weight. 5) Obtain W_d by subtracting weight of foil. Prior to using this procedure on the samples recovered from dissolution test units at Pantex and LLNL, the specimens were soaked overnight several times in clean DMSO to exchange any residual LX-10/DMSO solution with clean DMSO.

Aged Samples

It was previously noted that naturally aged lab specimens of 1977 vintage were obtained for testing.² In an attempt to simulate the additional postcuring and physical chemical changes that might occur over time in the stockpile, some of the Lab-Prepared Halthane 73-18 samples were subjected to accelerated thermal aging. These specimens were aged in a forced air oven at 50 and 70°C for durations up to 21 days.

Results & Discussion

Appendix A gives information on the composition and preparation of Halthane 73-18. As a familiarity with the microstructure of Halthane 73-18 will aid in subsequent discussions, a conceptual diagram of the network structure is shown in Figure 1.³ Halthane 73-18 is a "segmented" polyurethane, meaning that it is characterized by a morphology containing microphases of "hard" glassy segments and "soft" flexible chain segments. Elastomers with this type of morphology exhibit properties typical of crosslinked elastomers because the hard segments function as tie-points (or crosslinks) for the flexible soft segments. If crosslinking chemistry is not utilized, this segmented morphology also provides unique processability since the hard segments can be melted to yield a reformable thermoplastic polymer. Halthane 73-18, however, is crosslinked and thus it retains good network properties even at temperatures above the melting point of the hard segments.

³ L R Althouse and N W Hetherington, "Study of the Aging processes in Polyurethane Adhesives Using Thermal Treatment and Differential Calorimetric, Dielectric, and Mechanical Techniques. I. Identifying the Aging Process. II. Quantifying the Aging Effect," LLNL, UCRL-52849, November, 1979.

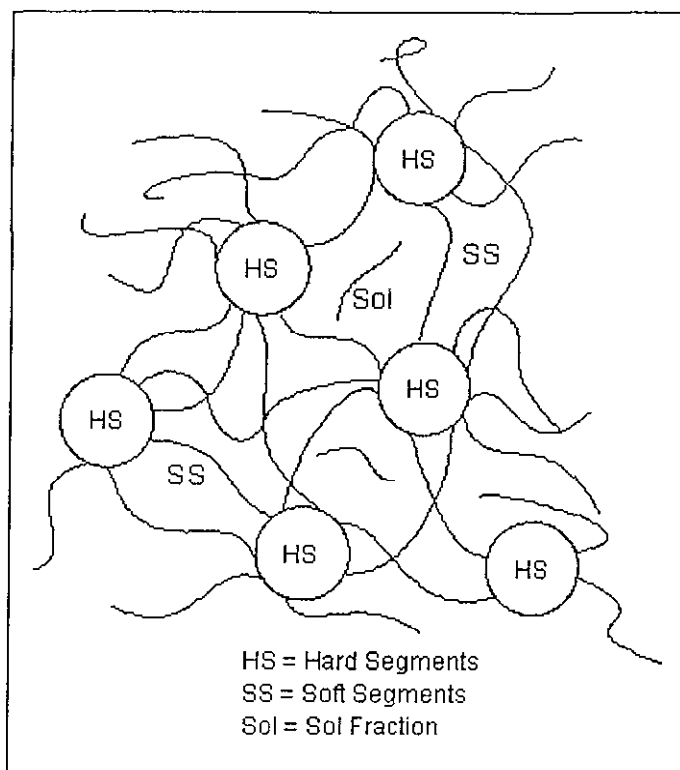


Figure 1 Segmented polyurethane structure comprised of hard and soft segment domains

Unattached network material, or "sol," is shown in Figure 1. This material can have several origins, for example, off-stoichiometry cure chemistry (e.g., from inaccurate weighing, or incomplete mixing), reactions with atmospheric contaminants (especially water), or reactions with species on the surface of a substrate (if, say, the material is used as an adhesive). If "sol" is present in Halthane 73-18 then it may be extracted during DMSO swelling/drying experiments. The presence of sol may be important because it implies that the network does not contain the maximum concentration of crosslinks. This coupled with the extraction of the sol by a solvent may render the network weakened and possibly more susceptible to mechanical damage, say by the action of DMSO spray.

If Halthane is prepared under strictly controlled conditions with identical starting materials and similar cure conditions, then different batches should exhibit nearly identical morphologies and network characteristics. We would expect that such batches would also exhibit consistent swelling behavior. Our findings, however, revealed a situation that was far removed from this ideal. Figure 2 shows swelling results (as percent weight gain) for various *Retention* samples of Halthane 73 adhesive. These samples were cured at ambient (room temperature), and several were subjected to elevated temperature postcures. We included several Halthane 73-19 samples in this study because this adhesive is identical in composition to 73-18, except for the addition of a catalyst to accelerate the cure.⁴

DMSO-induced swelling of Halthane 73 adhesives appears to span a wide range of weight gains (130-360 wt%). This cannot be attributed to rate effects, as the data clearly shows that equilibrium swelling is essentially attained in these small specimens after just 24 hours of immersion in DMSO at 50°C. The

⁴ H G Hammon, L P Althouse, and D M Hoffman, "Development of Halthane Adhesives for Phase 3 Weapons Summary Report," LLNL, UCRL-52943, December 1980

data in Figure 2 also shows that the swelling behavior does not correlate with sample age. Indeed, in two cases, multiple batches made on the same day exhibited significantly different swelling behaviors (20-Oct-93 and 5-Nov-93 specimens). Finally, the data also suggests that postcuring has little effect on the swelling behavior (the two 1-Oct-93 Halothane 73-19 samples). We will revisit the apparent effects of postcuring on swelling later in this section.

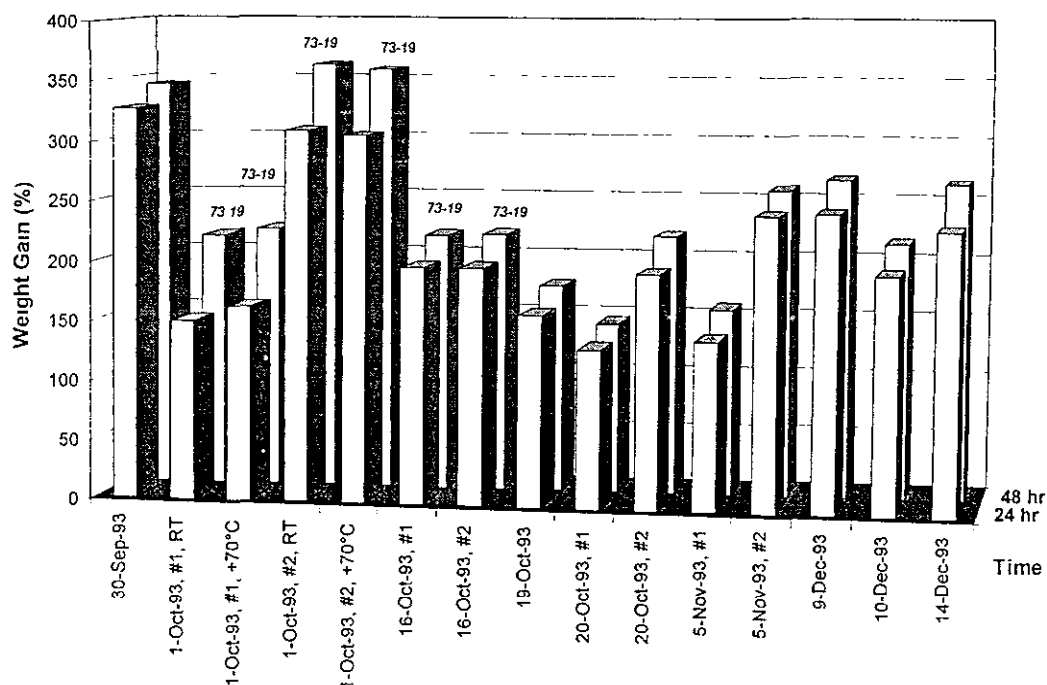


Figure 2 Swelling results for various Halothane 73-18 Retention specimens as function of immersion time. (The Halothane 73-19 labeled data is discussed in the text.)

Figure 3 and Table 1 show swelling data for four Halothane 73-18 films recovered from W79 units processed in the Pantex and LLNL Workstations. Also shown are swelling data for a half dozen lab-prepared Halothane specimens. As previously noted, it is unfortunate that weight gain and sol fraction values could not be obtained from the recovered workstation films because their initial dry weight is unknown. We will now discuss four observations from the data in Figure 3 and Table I.

First, the Type 6B films recovered from the Pantex and LLNL Workstations appear to be the least swollen of all the films evaluated (i.e., lowest weight loss on drying). The two films recovered from the LLNL-built W79 units (DTU79-3, -5) do not appear to be significantly different from the various lab-prepared samples. The results for the Type 6B films could be interpreted as a confirmation of Pantex's experience with Type 6B units, i.e., that the Halothane is more difficult to eliminate. We are hesitant, however, to arrive at this conclusion based on such a limited set of data, especially since the W79 Type 6B unit evaluated in the LLNL Workstation did not prove to be any more difficult to process than the LLNL-built units. The practical issue, however, is how easily the films are destroyed by the action of the DMSO spray. Although we never recovered very large pieces of swollen films, our simple handling tests (with forceps) did not distinguish any revealing differences in the appearance, strength or toughness of the films. Our primary observation from these handling experiments was that all the films were very easy to tear or shred once nicked.

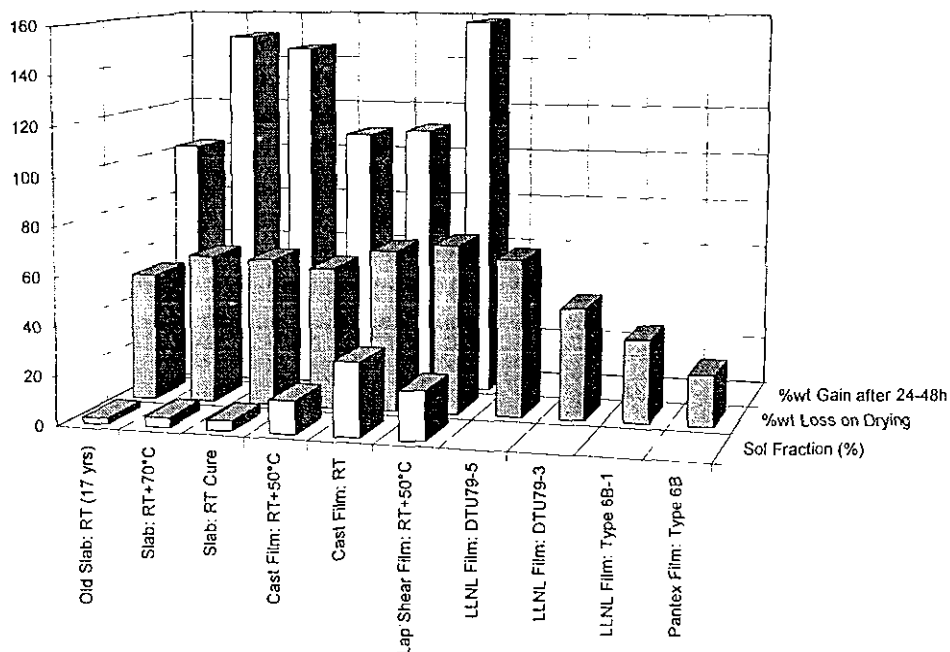


Figure 3 Swelling results on various Halothane 73-18 specimens The four specimens on the right were recovered from actual W79 dissolution test runs

Table I Swelling results on various Halothane 73-18 specimens

Sample Description	Gain (wt%)	Loss (wt%)	Sol (wt%)	Source
Old Slab 17years @ R T	102	52.7	2.5	Y-12
Slab R T Cure	146	60.9	3.8	B222
Slab R T + 21 days @ 70°C	151	61.5	3.8	B222
Cast Film R T Cure	111	66.4	29.6	B222
Cast Film R T + 7 days @ 50°C	109	58.4	13.1	B222
Lap-shear Film R T + 4 days @ 50°C	158	69.3	19.6	S300
Recovered Film, LLNL DTU79-5		64.4		S300
Recovered Film, LLNL DTU79-3		45.7		S300
Recovered Film, LLNL Type 6B-1		34.0		S300
Recovered Film, PX Type 6B		20.8		Pantex

R T = Room Temperature

Our second observation from the data in Figure 3 is that aging, whether naturally or thermally induced, appears to have no significant effect on the swelling behavior of Halothane 73-18. Comparing the DMSO swelling of the 17-year old Y12 slab specimen to the LLNL-prepared slab suggests that reduced swelling might accompany aging, however, laboratory aging of a lab-prepared slab at modest temperatures failed to reproduce this result. On the other hand, results for the Cast Film in Table I, show no significant change in swelling, but there is evidence that the amount of sol is reduced after aging. For this reason we carried out additional experiments to determine the effects of elevated temperature postcuring/aging on bonded films of Halothane 73-18.

Figures 4 and 5 show the effects of 50 and 70°C elevated temperature exposure, or postcure, on both the swelling behavior and bond strength of thin adhesive films of Halthane 73-18 prepared in a lap-shear configuration on aluminum adherends. The swelling tests were performed on the adhesive films recovered from the lap-shear specimens after testing. The range of the weight gains measured for these samples (120-230 wt%) were consistent with the values shown in Figures 1 and 3. The data in Figures 4 and 5 show conclusively that postcuring has no effect on the lap-shear bond strength and swelling behavior of Halthane 73-18. Increased bond strength and reduced swelling are two results we would expect to see if postcuring was causing additional curing or crosslinking. Conversely, if postcuring was causing degradation of the Halthane network, we would expect to see reduced bond strength and increased swelling. Neither of these trends are clearly exhibited by the experimental data.

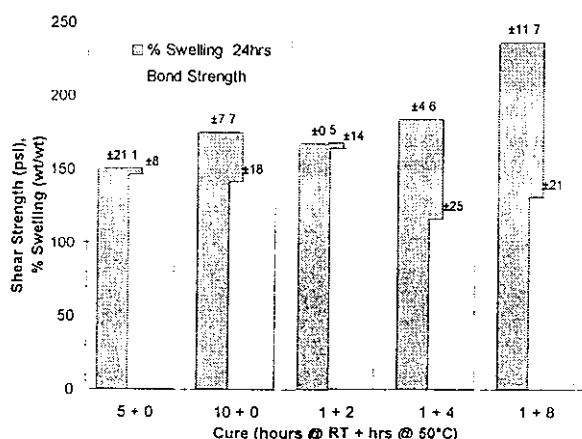


Figure 4 Halthane 73-18 lap shear strength and weight gain in DMSO as function of cure at room temperature and postcure at 50°C (Note Weight loss on drying and sol fraction data for the "1+4" film is shown in Figure 3 and Table I)

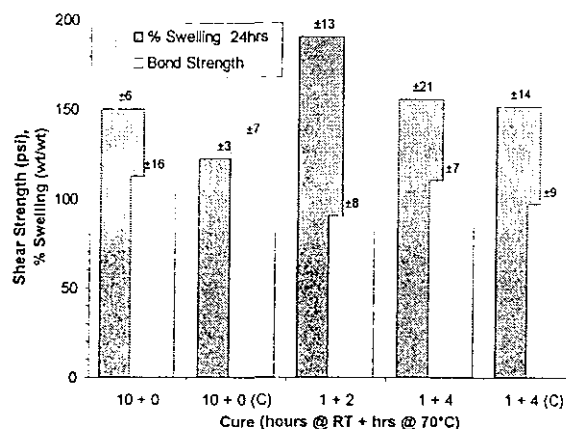


Figure 5 Halthane 73-18 lap shear strength and weight gain in DMSO as function of cure at Room temperature and postcure at 70°C ("C" denotes degassed batches of adhesives)

Returning again to Figure 3, our third observation is that larger sol fractions are observed in the films than in the bulk samples in Figure 3. One reasonable explanation for this is that the high surface-to-volume ratio of films makes them more susceptible to contamination by atmospheric and adsorbed surface moisture. For example, adsorbed water could react with the isocyanate functional groups and reduce the curing and crosslinking functionality. This, in turn, could result in an increased sol fraction because pre-reacted molecules cannot be fully incorporated into the network structure.

Our fourth and final observation from Figure 3 is that there is a weak correlation between weight gain on swelling and weight loss on the drying of swollen Halthane 73-18 (Figure 6). The curves plotted in Figure 6 show the relationship between weight gain and weight loss as a function of sol fraction (Equation 4).

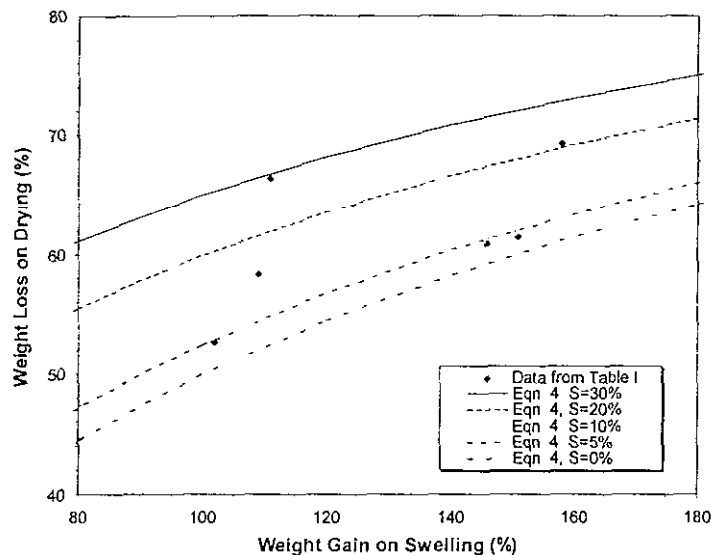


Figure 6 Correlation between weight gain on swelling and weight loss on drying for samples listed in Table I. The curves represent Equation 4 with sol fractions (S) ranging from 0-30 wt%

Conclusions

No definitive chemical or aging phenomena were identified to explain the differences observed in the behavior of DMSO-swollen films of Halthane 73-18 in the LLNL and Pantex HE Dissolution Workstations. The most likely explanation would appear to be that the less vigorous DMSO spray conditions in the Pantex Workstation were insufficient to shred and eliminate the swollen Halthane film.

If a material property of Halthane was a factor for the observed difference in behavior, we can only speculate that might be due a difference in sol fractions in the adhesive films. The argument would be that that different sol fractions might influence the ease with which swollen films of Halthane could be torn apart under the action of DMSO spray. The presence of a large fraction of sol in a Halthane 73-18 network could indicate poor curing and incomplete crosslinking, which might weaken the polymeric network and make it easier to destroy. In thin adhesive films it is possible that adsorbed water on the bonding surface could interfere with the urethane curing chemistry. In the few measurements made in this study, thin films of lab-prepared Halthane 73-18 did indeed exhibit larger sol fractions than did bulk samples.

Mechanisms by which it might be possible for the W79 Type 6B units to have smaller sol fractions than the LLNL-built units can only be speculated. One possibility is that the sol fraction initially present in the older units was been chemically incorporated back into the network, perhaps by interaction with compounds and degradation products from the LX-10 high explosive. However, the most obvious reaction, oxidation by NO_2 from the HMX in LX-10, would seem more likely to result in chain scission rather than addition, which would increase rather than decrease the sol fraction. Additional experiments would need to be conducted to determine if Halthane adhesive films are indeed altered by prolonged contact with LX-10.

Acknowledgments

Thanks go to Ing Chiu for conducting the experimental work and George Dorsey at Y-12 for providing samples of aged Halthane 73-18.

Appendix A

Halthane 73-18 Information

Composition:

- PART "A" (Halthane 73)

PolyMeg 1000	47.6 wt%	Polyol
PolyMeg 2000	7.4 wt%	Polyol
MDI	45.0 wt%	Diisocyanate

- PART "B" (HGH-18)

PolyMeg 1000	85 wt%	Polyol
Butanediol	10 wt%	Diol chain extended
Quadrol	5 wt%	Tetrafunctional crosslinker

Preparation:

- Add 35 parts by weight of Part "B" to 65 parts of Part "A" and mix

Curing:

- Pot Life 72 min (100 grams)
- Handling time 16 hrs

Special Notes:

- HGH-18 crystallizes on setting. Before use, crystallized material should be heated to ~50°C to dissolve the crystals. The material must be cooled back to room temperature before use.
- Cure at room temperature cure produces opaque white material
- Cure at 100°C produces transparent material
