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CHARACTERIZATION OF INTERGRANULAR PHASES IN TETRAGONAL AND CUBIC YTTRIA-STABILIZED ZIRCONIA

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Achieving superplasticity in fine-grained ceramics is a potential method to lower energy costs associated with ceramic manufacturing via net shape forming. Superplasticity is intrinsic in 3-mol%-yttria-stabilized tetragonal zirconia polycrystals (3Y-TZP), and can be enhanced by addition of glass to form intergranular phases which are thought to both limit grain growth and promote grain boundary sliding during processing (sintering and hot isostatic pressing).^{1,2} This permits processing at lower temperatures. However, superplasticity has not been observed in 8-mol%-yttria-stabilized cubic zirconia (8Y-CSZ), ostensibly due to its larger grain size and high grain growth rates.^{3,4} As part of a larger study, high-spatial-resolution energy-dispersive X-ray spectrometry (EDS) has been performed on 3Y-TZP and 8Y-CSZ specimens doped with various glassy phases to characterize intergranular compositions.

Zirconia powders were mixed with glass to produce specimens having either 1 wt % lithium-aluminum-silicate, 1 wt % barium-silicate, or 1 wt % borosilicate. Some specimens were prepared without added glass. Following sintering and hot isostatic pressing, specimens were prepared for transmission electron microscopy by conventional grinding, polishing, and ion-milling techniques. Specimens were mounted on copper slotted washers and carbon-coated prior to examination.

An EMiSPEC Vision integrated acquisition system interfaced to a Philips CM200FEG with an Oxford super-ATW detector and XP3 pulse processor was used to acquire EDS spectrum lines (typically 20 points, 0.5 to 2.0 nm spacing, 10 s dwell/point) across edge-on grain boundaries and across grain-boundary triple-points. Spectra were "post processed" interactively with the Vision software to yield profiles of background-subtracted integrated intensities. Additionally, some spectra were acquired for 100 s live-time with probes which were either stationary or rastered (to reduce beam damage) within the region of interest.

The grain diameters of 8Y-CSZ (Fig. 1) and 3Y-TZP (Fig. 2) specimens were approximately 4 μm and 200 nm, respectively. Spectra were acquired from regions marked in these figures. These spectra indicate that in both undoped zirconias yttrium levels increase and zirconium levels decrease at grain boundaries and triple points. Furthermore, profiles acquired across doped 8Y-CSZ grain boundaries always indicated the presence of an intergranular phase (Figs. 1b and 3). It is interesting to note here that the yttrium $K\alpha$ signal is constant across the boundary depicted in Fig. 3. The Ba and Si profiles have a full-width-half-maximum (FWHM) of ~ 2 nm, as does the Zr depletion profile, and indicate that the thickness of the intergranular phase (or "chemical width" of the boundary) is less than this since the probe diameter was also ~ 2 nm (with ~ 1.5 nA). To reduce beam damage (noted in Fig. 1b) and in an attempt to improve spatial resolution, 3Y-TZP specimens were examined with a finer probe (1.2 nm FWHM; 0.5 nA). Initial EDS results indicate full coverage of zirconia grains by the glassy phase may not be achieved. This is likely as the smaller grain size of 3Y-TZP corresponds to more grain boundary per volume over which the glassy phase spreads. However, the reduced EDS signal warrants examination of these boundaries with higher probe currents. Spectra acquired with stationary probes (Fig. 4) do reveal glassy phase at triple points and at some locations along grain boundaries. Further characterization of intergranular phases in these ceramics is planned to help achieve improved superplasticity through structure-processing-property correlations.⁵

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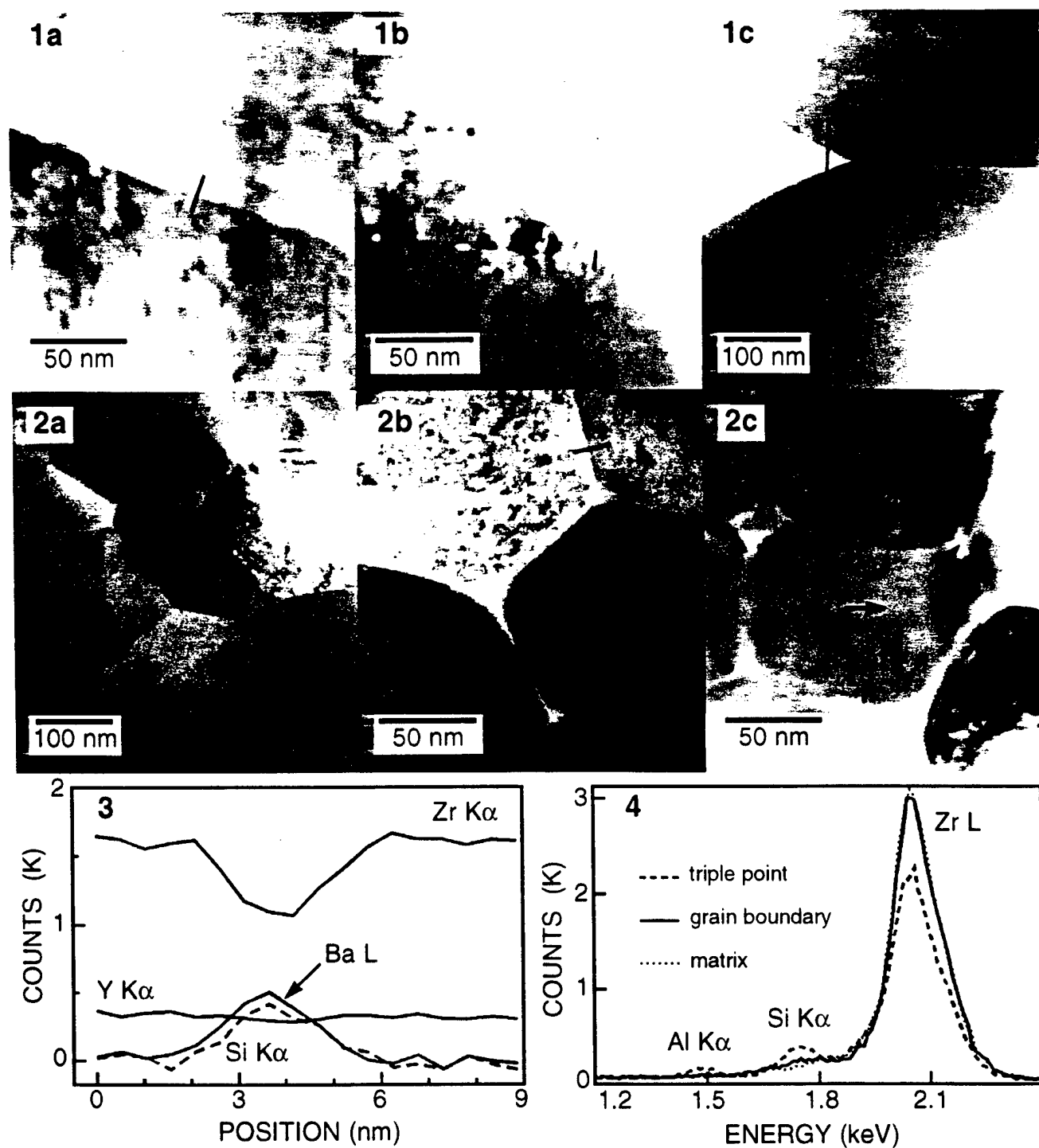


FIG. 1 - 8Y-CSZ (a) undoped, (b) barium-silicate doped, (c) lithium-aluminum-silicate doped.
 FIG. 2 - 3Y-TZP (a) undoped, (b) barium-silicate doped, (c) lithium-aluminum-silicate doped.
 FIG. 3 - EDS profiles (10 s dwell per pixel) acquired across grain boundary identified in Fig. 1b.
 FIG. 4 - EDS spectra (100 s dwell, stationary probe) from locations identified in Fig. 2c.

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