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## **Chemical and Sonochemical Approaches to the Formation of VO<sub>2</sub> and VO<sub>2</sub>-Impregnated Materials**

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# CHEMICAL AND SONOCHEMICAL APPROACHES TO THE FORMATION OF VO<sub>2</sub> FILMS AND VO<sub>2</sub>-IMPREGNATED MATERIALS

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## ABSTRACT

A new chemical and chemical/ultrasonic approach to the preparation of VO<sub>2</sub> films and VO<sub>2</sub>-impregnated bulk materials has been developed. In this approach, a V<sub>2</sub>O<sub>5</sub> sol prepared by quenching is used to coat SiO<sub>2</sub> substrates. The resulting gel-film is heat treated in a reducing atmosphere to form a film identified as VO<sub>2</sub> from the results of X-ray diffraction and both optical and resistivity measurements, which reveal the phase transition characteristic of vanadium dioxide. The advantage of this approach to the formation of VO<sub>2</sub> is that the V<sub>2</sub>O<sub>5</sub> sol can be used to impregnate porous materials, which are then heat treated to form an optically active composite material. The switching properties of the VO<sub>2</sub> films are investigated using optical and resistivity measurements, and the results are compared to those obtained for VO<sub>2</sub>-films prepared by more-conventional methods.

## INTRODUCTION

Vanadium dioxide, both in single-crystal form and as a thin film, has been the subject of a number of previous investigations. These prior studies were motivated by the large variations in the electronic and optical properties of VO<sub>2</sub> that occur as the material undergoes a structural phase transition [1]. This transition is also accompanied by a semiconductor-to-metal transition that is characterized by a change in the VO<sub>2</sub> electrical conductivity of over two orders of magnitude. The phase-transition temperature of ~68°C is sufficiently close to ambient temperature to make this material suitable for a variety of practical applications, including thermally activated optical switching and data storage devices [2,3], modulators and polarizers of submillimeter wave radiation [4], and energy-conserving coatings for windows and walls [5]. Single crystals were found to disintegrate upon cycling through the transition temperature; thin films, however, appear to survive the stresses during repetitive cycles and are, therefore, considered much more practical for applications. High-quality VO<sub>2</sub> films have been obtained using vapor deposition [6], metal oxidation and reactive evaporation [7], reactive sputtering [8], pulsed laser deposition [9], and both organic and inorganic sol-gel methods [10,11,12]. Recently, optically active switchable surfaces have been produced by ion-implantation: by co-implanting vanadium and oxygen into a sapphire substrate and annealing under the proper conditions, it was possible to form buried precipitates of VO<sub>2</sub> [13]. The advantage of this type of active composite surface arises from the fact that the active components are an integral part of the near-surface region of the host material and are, therefore, protected from potential environmental degradation effects.

In this work, a new chemical and chemical/ultrasonic approach to the preparation of VO<sub>2</sub> films and VO<sub>2</sub>-impregnated bulk materials has been developed. In this approach, a V<sub>2</sub>O<sub>5</sub> sol prepared by quenching is used to coat SiO<sub>2</sub> substrates. The resulting gel-film is heat treated in a reducing atmosphere to form a film identified as VO<sub>2</sub>, revealing the switching behavior characteristic of vanadium dioxide. The fact that this approach is a relatively simple and inexpensive way to form VO<sub>2</sub> films makes it very valuable. Its main benefit, however, may lie in

the possibility of using the  $V_2O_5$  sol to impregnate porous materials, which can then be heat treated to form an optically active composite material, consisting of  $VO_2$  nanocrystals embedded in the host matrix. Potential candidates for host materials can be found among the so-called sonogels. In contrast to "classic" sol-gels, a sonogel is made without the addition of a solvent. Instead, the different compounds are mixed using ultrasonic energy. Ultrasonic waves propagating through a liquid can generate locally a large amount of energy through the collapse of cavitation bubbles [14]. It has been demonstrated that homogeneous solutions can be formed when 2 immiscible liquids are simultaneously subjected to ultrasound radiation, as a result of the forces acting at their interface. The sonicated water-TEOS solutions quickly gel, giving rise to the so-called sonogels with a superior short gelation time, high density, and a fine, uniform porosity [15]. The surface:volume ratio is typically twice as large as that for gels prepared in an alcoholic solution, which permits sintering at lower temperatures [16]. Sonogels are considered to be superior hosts for nanoparticles and dyes, producing materials with nonlinear optical applications, as well as ceramic composites [17]. The objective of the research effort reported in this paper is to form an optically active composite material, consisting of  $VO_2$  nanocrystals embedded in a silica sonogel.

## EXPERIMENTAL DETAILS

In the subject synthesis process, 15 grams of  $V_2O_5$  powder (99.995%, Alfa Aesar, Puratronic) were put in a platinum crucible and melted in air at 1100 °C for 20 minutes. The melt was then poured in 500 ml of distilled water and vigorously stirred. After discarding the residue on the bottom of the jar, a brown sol with approximately 1.6 wt.%  $V_2O_5$  was obtained. Thin films of  $V_2O_5$  were deposited on amorphous  $SiO_2$  substrates (25 x 10 x 2 mm) by dipping the substrates in the sol. A 2-step recipe was used to reduce the  $V_2O_5$  film to  $VO_2$ . The first step consists of reducing the  $V_2O_5$  film at 500 °C in argon gas, containing 4 % hydrogen. Under these conditions, the brown film turns gray and converts to mainly  $V_2O_3$ . In order to obtain  $VO_2$ , a second heat treatment is carried out at 450 °C in argon gas, bubbling through water. The water vapor treatment provides a slightly oxidizing atmosphere, which leads to the formation of a stable blue-colored  $VO_2$  film.

The preparation of the silica sonogels is based on the method of De La Rosa-Fox et al. [15]. In this approach, sonogels are obtained by submitting mixtures of  $Si(OCH_3)_4$  (TEOS),  $H_2O$  (pH=1) and formamide in a molar ratio 1:10:7 to ultrasonic radiation, produced by a sonifier (Misonix, Model XL2015). Gelling and aging took place at 40 °C.

## RESULTS AND DISCUSSION

Figure 1 shows an X-ray  $\theta$ - $2\theta$  scan (using Cu  $K\alpha$  radiation) of a vanadium oxide film formed on a  $SiO_2$  substrate, after being reduced under the conditions described above. The intense peak observed at 27.86° is characteristic of the (011) reflection of  $VO_2$  [18], and suggests, together with the reflections at higher angles, that the film consists mainly of vanadium dioxide. There is, however, at least one impurity phase present, which is difficult to identify because of the large number of nonstoichiometric vanadium oxides that can be formed. Since these oxides could have reflections close to the reflections expected for vanadium dioxide, resistance measurements as well as optical transmission measurements, have been performed in order to confirm unambiguously the presence of  $VO_2$ . Figure 2 shows the resistance as a function of temperature for the  $VO_2$  film on the  $SiO_2$  substrate, as the temperature is cycled from room temperature to 100 °C. This figure clearly shows a drastic change in the resistance of the film upon heating and cooling in the 65°-70° C region, i.e. the temperature region where the characteristic  $VO_2$ -phase

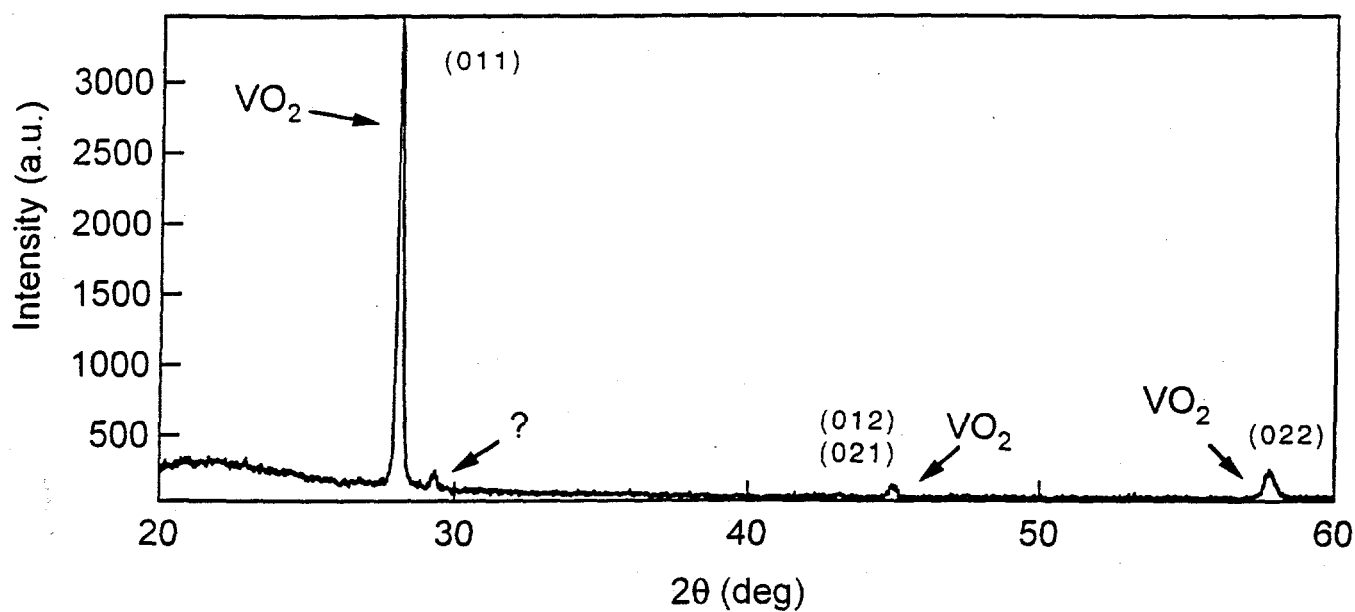


FIGURE 1: X-ray diffraction pattern of a  $\text{VO}_2$  film obtained by heat treatment of a  $\text{V}_2\text{O}_5$  gel-film.

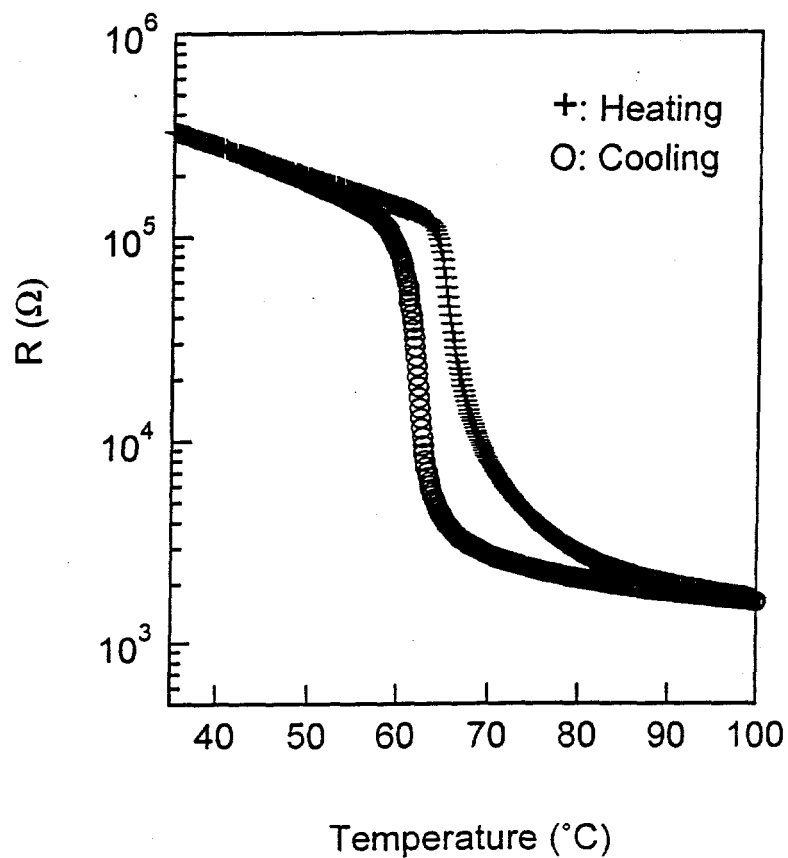


FIGURE 2: Resistance as a function of temperature for a  $\text{VO}_2$  film obtained by heat treatment of a  $\text{V}_2\text{O}_5$  gel-film.

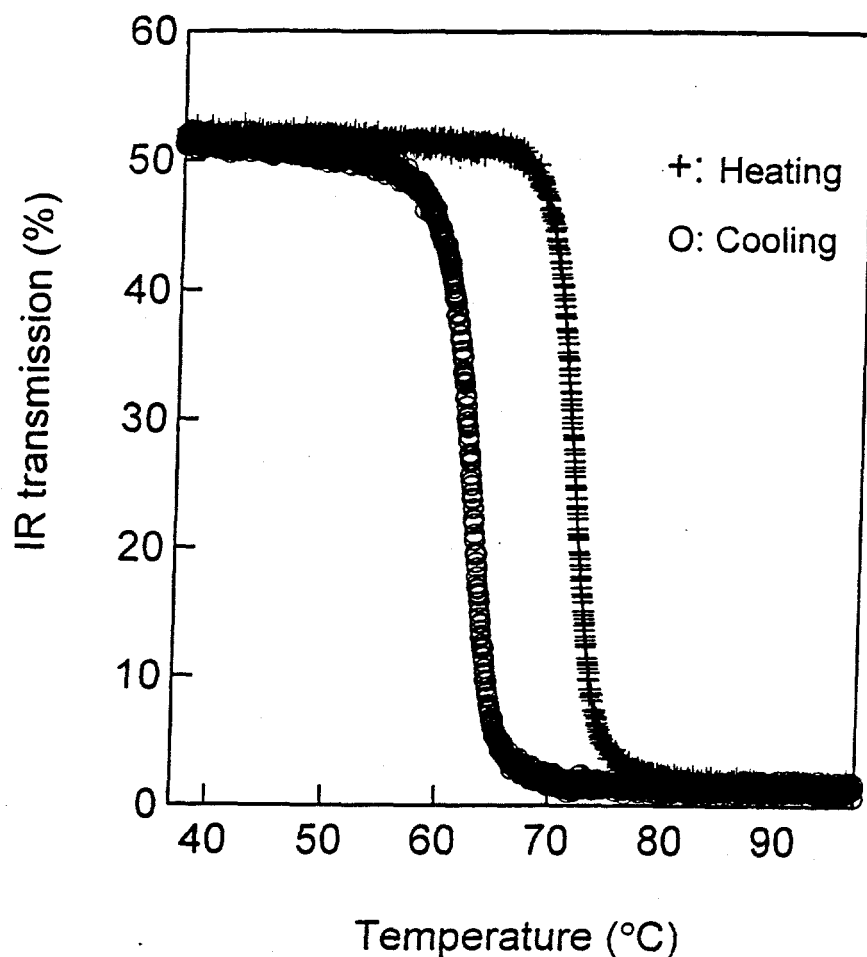


FIGURE 3: IR-transmission ( $\lambda = 3.4\mu\text{m}$ ) as a function of temperature for a  $\text{VO}_2$  film obtained by heat treatment of a  $\text{V}_2\text{O}_5$  gel-film. The thickness of the film is approximately  $4500 \text{ \AA}$ .

transition takes place. A change of more than 2 orders of magnitude is observed in our film at a transition temperature of  $68^\circ\text{C}$ , which is typical for films deposited on glass substrates. Although the magnitude of resistive switching has been reported to be between 3 to 5 orders of magnitude for single crystals or thin films on crystalline substrates, it is no more than 2 to 3 orders of magnitude for films on glass substrates [19]. Figure 3 shows the optical transmission for the film, measured at a fixed wavelength of  $3.4 \mu\text{m}$ . It clearly illustrates the drastic change in transmission as the film is cycled through the transition region. Not only is the switching contrast comparable to values reported for "standard" films [20,21], the transition is also remarkably sharp.

The results of this sol-gel approach, as established by the properties of the resulting films, are extremely promising and indicate that superior switching properties may be obtained when the coating-procedure is optimized and the influence of the film-thickness and length of the heat treatments is determined. It will also be interesting to investigate the properties of films deposited on crystalline substrates. However, as noted above, the main benefit of this approach may be the possibility of using a  $\text{V}_2\text{O}_5$  sol to impregnate porous materials and form thereby "switchable" composite materials. To investigate these possibilities, we have originated a study of silica

sonogels as possible hosts for  $\text{VO}_2$  nanocrystals. Sonogels were preferred over conventional gels because of their short gelation time and their excellent properties as a host matrix for nanoparticles [17]. One possible way of making such a  $\text{SiO}_2/\text{VO}_2$  composite is an "in situ" impregnation, where that the  $\text{V}_2\text{O}_5$  sol can be directly used as one of the compounds to make the sonogel. We were, in fact, able to form homogeneous monolithic gels by adding a  $\text{V}_2\text{O}_5$  sol to the TEOS-water-formamide mixture. However, in this approach, one has to deal with the fact that  $\text{V}_2\text{O}_5$  dissolves in formamide, which is used as a drying-control chemical additive (DCCA). Currently, attempts are being made to form homogeneous  $\text{V}_2\text{O}_5$ -containing gels omitting the DCCA or by using another drying-control agent. A second approach to the formation of  $\text{SiO}_2/\text{VO}_2$  composites consists of "infiltrating" the  $\text{SiO}_2$  after gelling. In order to avoid having the  $\text{V}_2\text{O}_5$  go into solution, the gel-glass has to be sufficiently dry, such that most of the formamide has been removed. Successful infiltration was obtained by soaking the gels, which had been dried at  $250^\circ\text{C}$ , in a diluted  $\text{V}_2\text{O}_5$  sol. These gels have been subjected to the heat treatments that were found to give the proper reducing condition for the  $\text{VO}_2$ -films. The first heat treatment, at  $500^\circ\text{C}$  in an  $\text{Ar}/4\%\text{H}_2$  atmosphere changes the color of the film from yellow to gray. This color and the observation that parts of the gel are conducting indicate that the  $\text{V}_2\text{O}_5$  in the gel has, like the  $\text{VO}_2$  films, been reduced to  $\text{V}_2\text{O}_3$ . The second heat treatment, carried out at  $450^\circ\text{C}$  in Argon bubbling through  $\text{H}_2\text{O}$ , returns the gel to its original yellowish color, and no indication of a phase transition around  $68^\circ\text{C}$  could be observed. It is likely that the slightly oxidizing atmosphere, provided by the water vapor, does not supply the right atmosphere for the gel/ $\text{V}_2\text{O}_3$  system, probably because of the water that is still retained or absorbed by the  $\text{SiO}_2$  gel. As a result, a higher oxide is obtained instead of the intended  $\text{VO}_2$ . Heat treatments under different conditions are currently being carried out in order to find the proper conditions to form a stable  $\text{SiO}_2/\text{VO}_2$  composite.

## CONCLUSIONS

A new chemical and chemical/ultrasonic approach to the preparation of  $\text{VO}_2$  films and  $\text{VO}_2$ -impregnated bulk materials has been developed. In this approach a  $\text{V}_2\text{O}_5$  sol prepared by quenching is used to coat  $\text{SiO}_2$  substrates. The resulting gel-film is heat treated in a reducing atmosphere to form a film identified as  $\text{VO}_2$ . Resistance and IR transmission measurements as a function of temperature reveal the switching behavior characteristic of vanadium dioxide and indicate that superior switching characteristics may be obtained when the method is optimized. The possibility of using the  $\text{V}_2\text{O}_5$  sol to impregnate porous materials, which can then be heat treated to form an optically active composite material, has been investigated using silica sonogels as the host matrix. Partially dried gels have been infiltrated with the  $\text{V}_2\text{O}_5$  sol; however, the heat treatment conditions used to reduce  $\text{V}_2\text{O}_5$  gel films to  $\text{VO}_2$  do not seem to be adequate for the  $\text{SiO}_2$  gel/  $\text{V}_2\text{O}_5$  system. Future investigations will, therefore, focus on finding the proper reducing atmosphere to form a stable  $\text{SiO}_2/\text{VO}_2$  composite. Attention will also be paid to the formation and reduction of stable and homogeneous silica/  $\text{V}_2\text{O}_5$  composite gels, using the  $\text{V}_2\text{O}_5$  sol as one of the starting compounds for the sonogel.

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## REFERENCES

1. F. J. Morin, Phys. Rev. Lett. **3**, 34 (1959).
2. A. W. Smith, Appl. Phys. Lett. **23**, 437 (1973).
3. I. Balberg and S. Trokman, J. Appl. Phys. **46**, 2111 (1975).
4. J. C. C. Fan, H. R. Fetterman, F. J. Bachner, P. M. Zavracky, and C. D. Parker, Appl. Phys. Lett. **31**, 1 (1977).
5. C. V. Jorgenson and J. C. Lee, Solar energy Materials **14**, 205 (1986).
6. J. B. MacChesney, J. F. potter, and H. J. Guggenheim, J. Electrochem. Soc **115**, 52 (1968).
7. F. C. Case, Applied Optics **28**, 2731 (1989).
8. A. Razavi, T. Hughes, J. Antinovitch, and J. Hoffman, J. Vac. Sci. Technol. A **7**, 1310 (1989).
9. H. S. Choi, J. S. Ahn, J. H. Jung, T. W. Noh, and D. H. Kim, Phys. Rev. B **54**, 4621 (1996).
10. K. R. Speck, H. S.-W. Hu, M. E. Sherwin, and R. S. Potember, Thin Solid Films **165**, 317 (1988).
11. D. P. Partlow, S. R. Gurkovich, K. C. Radford, and L. J. Denes, J. Appl. Phys. **70**, 443 (1991).
12. Y. Dachuan, X. Niankan, Z. Jingyu, and Z. Xiulin, Mat. Res. Bull, **31**, 335 (1996); J. Phys. D **29**, 1051 (1996).
13. L. A. Gea and L. A. Boatner, Appl. Phys. Lett. **68**, 3081 (1996); L. A. Gea, L. A. Boatner, J. D. Budai, and R. A. Zuhr, in Ion-Solid Interactions for Materials Modification and Processing, edited by D. B. Poker, D. Ila, Y.-T. Cheng, L. R. Harriott, and T. W. Sigmon (Mat. Res. Soc. Proc. **396**, Pittsburgh, PA, 1996) pp. 215-220.
14. H. G. Flynn in Physical acoustics, edited by W. P. Mason (Academic Press, New York. 1964) **1B**, p57.
15. N. De La Rosa-Fox, L. Esquivias, and J. Zarzycki, J. Mater. Sci. Lett. **10**, 1237 (1991).
16. M. Pinero, M. Atik, and J. Zarzycki, J. Non-Cryst. Solids **174&148**, 523 (1992).
17. D. Levy and L. Esquivias, Adv. Mater. **7**, 120 (1995).
18. JCPDS-ICDD, 12 Campus Boulevard, Newton Square, PA 19073-3273.
19. C. B. Greenberg, Thin Solid Films, **110**, 73 (1982).
20. F. C. Case, J. Vac. Sci. Technol. **A8**, 1395 (1990).
21. E. E. Chain, Applied Optics **30**, 2782 (1991).