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DP-MS-79-30

CONF-800305--1
CONF-79112--46

CANISTER COMPATIBILITY WITH CARLSBAD SALT

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To be presented by
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Proposed for presentation at
1979 Annual Meeting
Materials Research Society
Cambridge, Mass.
November 26-30, 1979

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INTRODUCTION

This paper describes the results of the examination of candidate canister alloy specimens heated with Carlsbad salt in sealed capsules for up to 5000 hours and in unsealed capsules for up to 10,000 hours. The compatibility of candidate alloys, for use in fabricating canisters for solidified radioactive waste, with salt from a potential final storage location for waste forms is being evaluated by long-term heating tests.¹ In these tests, canister alloys are being exposed to salt in a manner similar to that expected during final storage of canisters of solidified waste forms in a salt deposit. Test specimens are being heated at the temperature expected during waste form storage and at higher temperatures to accelerate any reactions that may occur. Reaction rates are determined by examining replicate specimens periodically removed from the test.

SUMMARY

No significant reaction was found when candidate canister alloys were heated with salt from Carlsbad, New Mexico, for up to 5000 hours in sealed capsules and for up to 10,000 hours in unsealed capsules at temperatures (80 to 225°C) that bracket the maximum temperature calculated for reference Savannah River Plant (SRP) waste containers at 20-foot spacings in salt.² Additional tests were made at 600°C in sealed capsules to characterize reactions that may occur between candidate canister alloys and any component of the salt that is released when decrepitation occurs. Under these extreme conditions there was no significant attack of Type 304L stainless steel. But, there was up to 20-mils attack of the low-carbon steel.

DISCUSSION

Materials Used in Tests

Candidate Canister Alloys. Canister alloys for these tests were chosen for testing because of their oxidation resistance, low temperature ductility, and cost. Cor-Ten A (Cor-Ten is a trademark of U.S. Steel Company) is a low-alloy steel containing 1% chromium, 0.5% nickel, and 0.35% copper. This alloy costs only a little more than low-carbon steel, but it has better resistance to air oxidation.

ASTM-A-516 (Grade 70) covers a low-carbon steel plate that is suitable for use at low temperatures because of its low nil-ductility transition temperature (NDTT). This specification could be important if low-carbon steel is considered for use as the outer canister³ and low-temperature impact resistance is required as it is in the case of shipping casks for radioactive material.⁴

Past experience indicates that low-carbon and low-alloy steels have similar compatibility with vitrified waste. Therefore, the behavior of these two alloys should be representative of any other similar steels that might be selected.¹

Type 304L stainless steel has better resistance to air oxidation than either Cor-Ten A or low-carbon steel [ASTM-A-516 (Grade 70)]. It is more resistant than either of these alloys to accelerated corrosion in a radiation field. In a radiation field nitrogen combines with the oxygen and water vapor in the air to form nitric acid. But, stainless steel costs about three times more than the other alloys. It might not be suitable for final storage in salt because of its susceptibility to chloride pitting and/or stress corrosion cracking.

Carlsbad Salt. The salt used in these tests was from a section of core from Drill Hole AEC-8 in Carlsbad, New Mexico. The core, four inches in diameter and three feet long, was taken from the depth of 2142.7 to 2145.3 feet. Carlsbad is a potential final storage site for canisters of solidified radioactive waste and is being studied in the Waste Isolation Pilot Plant Program.^{5,6}

The salt is primarily halite (NaCl).⁷ When this salt was crushed, it gave off a faint odor suggestive of hydrogen sulfide. But, hydrogen sulfide was not confirmed by analyses of the salt.

As in most bedded (stratified) salt,⁸ moisture is trapped in small pockets on the grain boundaries and within the grains. When the salt is heated to around 250°C, the steam pressure in these pockets becomes greater than the bonding forces. This results in the salt fracturing with considerable violence. This

phenomenon, known as decrepitation, results in release of water vapor and hydrogen sulfide.

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Experimental Procedure

Two types of specimens are being used in these tests (1) unsealed capsules and (2) sealed capsules.

Unsealed Capsules. The first generation compatibility tests were started in unsealed capsules (Figure 1). Small capsules of Cor-Ten A and Type 304L stainless steel were tested. Low-carbon steel meeting specification ASTM-A-516 (Grade 70) was not used in these tests. These tests were started before the decision was made to test a material with properties meeting the low NDTT specification required for materials used in the construction of qualified shipping containers for radioactive material. In these tests the capsules were filled with crushed salt. The capsules were then covered, but not sealed, and placed in holes drilled in a solid piece of this salt. The entire assembly (salt block and capsule) was wrapped in aluminum foil in an attempt to confine any gases, such as hydrogen sulfide, or water vapor that might be present in the salt and that could influence the attack of the metal.

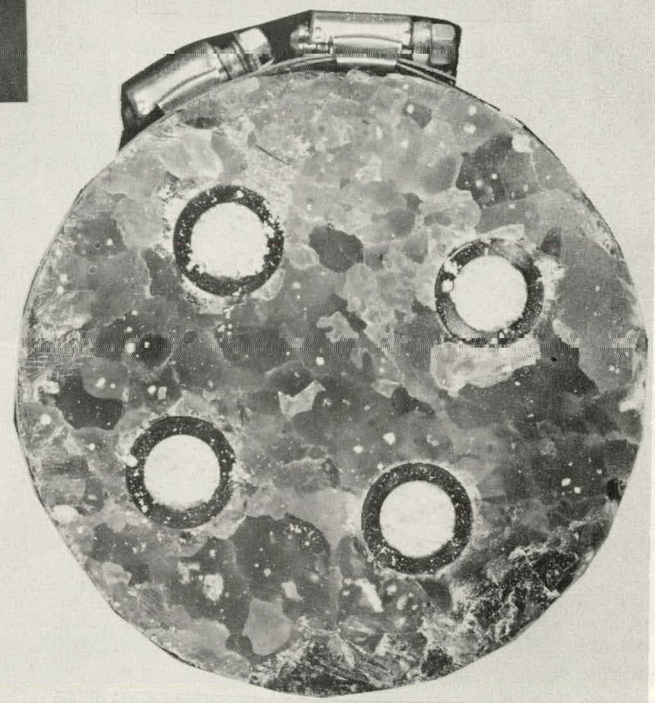
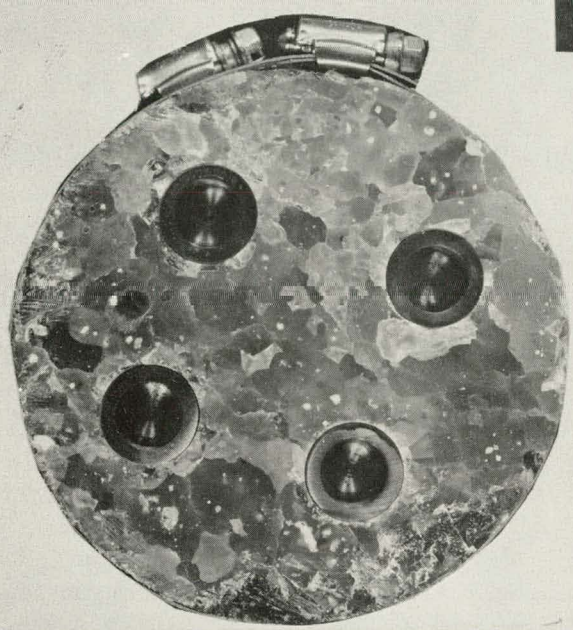
Unsealed capsules are being heated for up to 50,000 hours at 80°C, a reasonable temperature expected during waste form storage, and also at 225°C to accelerate any reactions that might occur. The maximum temperature of tests in unsealed capsules is limited to 225°C because the salt decrepitates at about 250°C. Decrepitation destroys this type of test assembly.

Sealed Capsules. The second series of compatibility tests were started in sealed capsules (Figure 2). Small right circular cylinders of low-carbon steel ASTM-A-516 (Grade 70), or Type 304L stainless steel were partially embedded in a small block of salt. This assembly was then put inside a capsule made of Type 304L stainless steel high-pressure pipe fittings. The capsules were sealed by tungsten-inert-gas (TIG) welding. These tests are designed to characterize reactions that occur between the canister alloys and any volatile components of the salt. Reaction of the portion of the specimen embedded in the salt will be indicative of the compatibility between the canister alloy and solid salt and vapor from the salt. Reaction of the portion of the specimen exposed only to the vapors from the salt will isolate the effect of the salt vapor above.

Specimens in sealed capsules are being heated for up to 50,000 hours at 80, 225, and 600°C. The amount of attack of the specimens heated in sealed capsules at 80 and 225°C will be compared with the amount of attack of specimens heated in unsealed capsules at these temperatures.

Test Capsule
3/4 in. dia.

Hose Clamps
to Keep Salt From
Breaking



Test Capsules in Holes
Drilled in Salt

Test Capsules Filled
with Crushed Salt

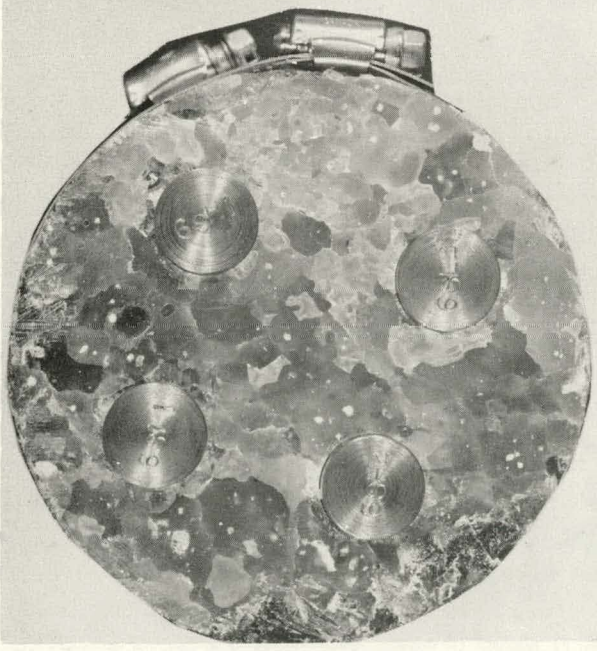


Fig. 1. Unsealed Capsule Assembly

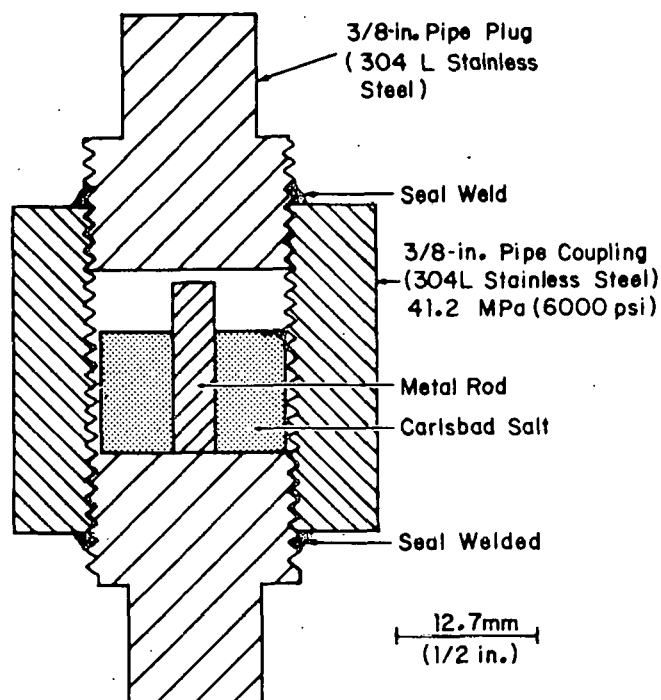


Fig. 2. Sealed Capsule Assembly

Tests at 600°C in sealed capsules were included in the test matrix to characterize any reactions that may occur between candidate canister alloys and any component of the salt that is released when decrepitation of the salt occurs. Decrepitation of the salt at about 250°C will occur, but this will not destroy the test assembly, as in the case of the unsealed capsules. The salt, the metal specimen, and any material that is released from the salt by decrepitation will remain in contact with the specimen inside the capsule.

Results

Unsealed Capsules. After the unsealed capsules had been heated for 10,000 hours, they were removed from the block of salt, the powdered salt was poured out of the capsules, and the surfaces of the capsules were examined visually and at low magnifications. There was no significant change in the appearance of the capsules heated at 80°C for 10,000 hours and the appearance of replicate specimens heated 1000 hours examined previously.⁹ Corrosion products and some pieces of salt adhered to the OD and ID surfaces of the Type 304L stainless steel capsule heated at 225°C for 10,000 hours. There was more corrosion products on the OD surface of the capsule than on the ID surface.

The dimensions of the unsealed capsules before and after heating were <5 mils which is within the experimental error of the technique used.

A cross section through capsules heated at 225°C was examined on the optical microscope and the scanning electron microscope (SEM). Very little attack was found of the Type 304L stainless steel capsule heated 10,000 hours at 225°C. No oxide film was found on the OD surface. Patches of oxide only 0.1 mil thick were found on the ID surface.

The ID surface of the low-carbon steel capsule heated 10,000 hours at 225°C was attacked less than the OD surface. An oxide film 0.6 mil thick was found on the OD surface. An oxide film 0.3 mil thick was found on the ID surface.

The capsules heated at 80°C were not examined on the microscope because no significant attack was found of capsules heated at 225°C.⁹

Sealed Capsules. The sealed capsules were opened by machining through the wall from the OD to the ID using a vertical milling machine. The capsule was not rotated so that damage to the salt and the specimen would be minimized.

The specimens heated at 80 and 225°C were pulled out of the block of salt with tweezers. The Type 304L stainless steel specimens showed no visible sign of oxidation. They had the same appearance as recently machined specimens. The low-carbon steel specimens were covered with a black oxide film. There was no difference in the appearance of either the end of the specimen exposed to the vapor or the end of the specimen in the hole in the block of salt.

The specimens heated at 600°C could not be pulled out of the fractured block of salt with pliers. The pieces of salt were chipped away with a small pick. Both ends of the Type 304L stainless steel specimen were covered with a thin black oxide film. But, no salt adhered to the specimen. The end of the low-carbon steel exposed to vapor was covered with a thick black film. Salt was bonded to the oxide on the end of the specimen in the salt.

No significant changes in dimensions occurred to specimens in test at conditions expected during waste form storage. Changes in diameter were <2 mils for Type 304L stainless steel and low-carbon steel specimens heated for up to 5000 hours at the temperature expected during waste form storage (80°C) and at a higher temperature (225°C) to accelerate any reactions that might occur.

In specimens exposed to more extreme conditions (600°C), there was no significant attack of Type 304L stainless steel in 5000 hours. But, extensive attack of the low-carbon steel occurred. The end of the specimen exposed to the vapor was attacked 20 mils in 5000 hours. The end of the specimen embedded in the salt was attacked 9 mils in 5000 hours.

The attack of the low-carbon steel specimen heated for 5000 hours with Carlsbad salt in a sealed capsule was characterized. X-ray diffraction analysis was performed on a piece of material from the surface of the portion of the specimen exposed to vapor. A longitudinal section through the entire specimen and transverse sections through each end of the specimen were examined on the SEM.

The appearance of the layers of material on both ends of the specimen indicate that the mechanism of attack of both ends was similar (Figure 3). There was about 0.8 mil of intergranular penetration of the low-carbon steel specimen. On the grain boundaries there was a material which contained iron and sulfur. Adjacent to the metal was a layer of material composed of the same elements as the low-carbon steel. This material was identified by x-ray diffraction analysis as iron oxide. The iron oxide layer was covered with a layer composed of iron oxide and a small amount of iron sulphide. Some particles of salt were embedded in this layer on the end of the specimen embedded in the salt.

A mechanism of accelerated corrosion could possibly be taking part in the oxidation of the low-carbon steel specimens in sealed capsules with Carlsbad salt at 600°C. Normally this reaction is associated with atmospheric corrosion in urban and industrial districts because of the presence of sulphur dioxide produced by the burning of fuel.^{10,11} First, the hydrogen sulfide from the decrepitated salt would react with the air inside the sealed capsule to form SO₃. The SO₃ would react with the water vapor from the salt forming H₂SO₄. The oxidation reaction of H₂SO₄ with the canister alloy liberates SO₃. This could lead to a cyclical series of reactions with iron where the reaction products are hydrated iron oxide and more sulfuric acid. This mechanism is consistent with the SEM, x-ray energy spectrometry (XES), and x-ray diffraction analysis of the films on the sample which showed that the specimen was covered with a layer of iron oxide and a layer of iron sulphide.

APPLICATION OF RESULTS

The results of these tests at 80 and 225°C show that there is no significant attack of Type 304L stainless steel or low-carbon steel under conditions expected during waste form storage and at a higher temperature designed to accelerate any reactions that might

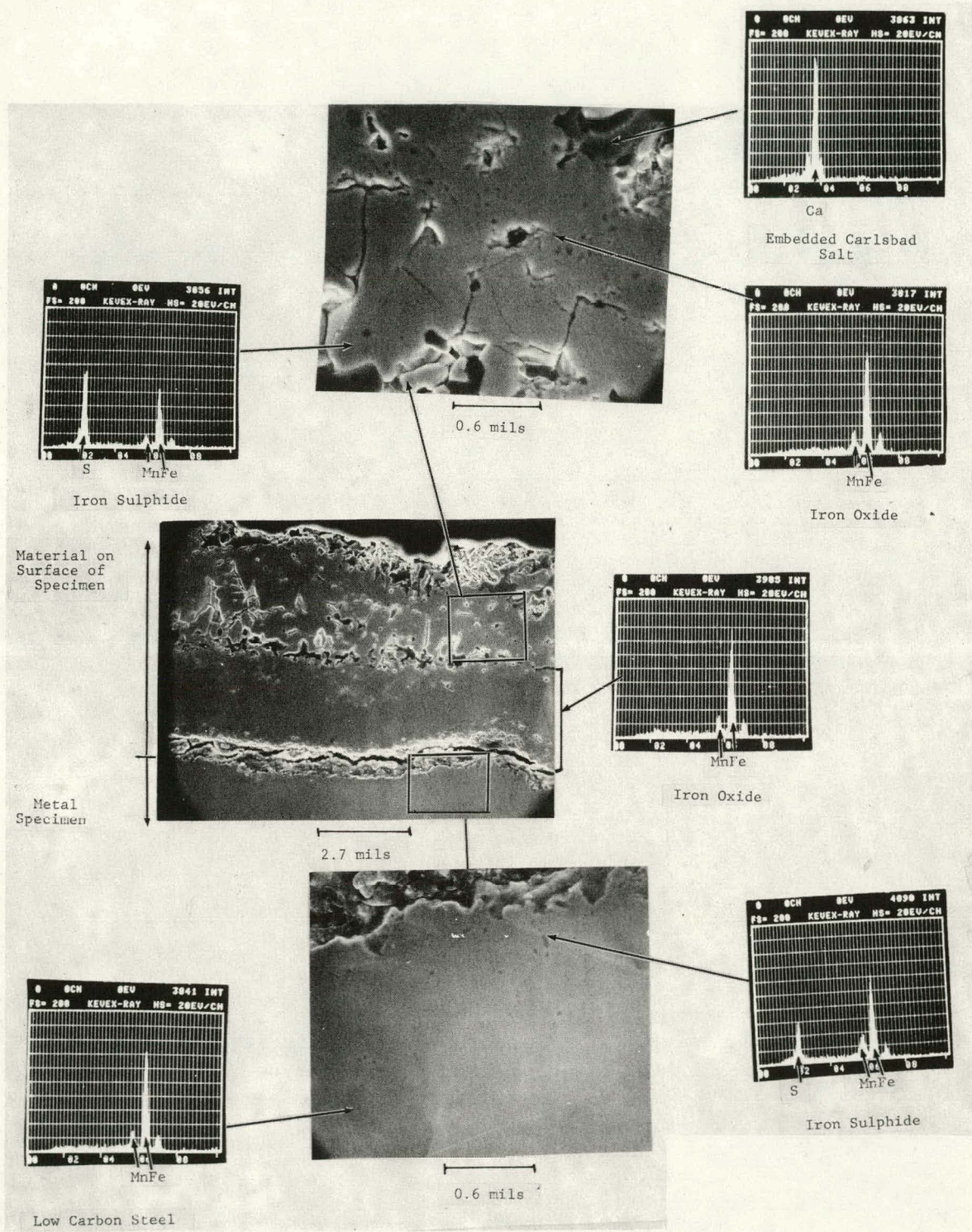


Fig. 3. Longitudinal Section Through the Surface of Low Carbon Steel Exposed to Carlsbad Salt in a Sealed Capsule (Penetration 9.3 mils in 5000 hours at 6000°C)

occur. The results of the tests at 600°C show that there is no significant attack of Type 304L stainless steel at this temperature by the volatile products of the salt released by decrepitation. But, there was extensive attack of the low-carbon steel under these extreme conditions. Storage conditions are expected to be designed to prevent temperatures high enough to cause decrepitation from occurring. However, because the canister of waste will be a higher temperature than the surrounding salt, the moisture trapped in small pockets on the grain boundaries and within the grains of the salt can migrate to the waste form during storage.¹² In addition, movement of this material may perhaps be enhanced by movement of the salt by creep to close in the space surrounding the waste form. Creep of the salt will occur because of the pressure of the overburden at this depth.

ACKNOWLEDGMENTS

The author is grateful to D. W. Powers of Sandia Laboratories for supplying the salt. He is also indebted to G. G. Wicks of Savannah River Laboratory (SRL) for the X-ray Diffraction Analysis and to J. A. Donovan formerly of SRL and P. K. Smith of SRL for their helpful discussions.

Work was done under USDOE Contract AT(07-2)-1.

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