

Version Date: 7/14/97

ICNS '97 Proceedings Paper to be published in Physica B- Condensed Matter

**Spectroscopic Study of the Proton Dynamics in Manganese Dioxide
Electrode Materials**

C. S. Johnson^a, M. M. Thackeray^a, J. C. Nipko^b, and C.-K. Loong^b

Argonne National Laboratory, Argonne, IL 60439, U.S.A.

^aElectrochemical Technology Program, Chemical Technology Division

^bIntense Pulsed Neutron Source Division

RECEIVED

AUG 26 1997

OSTI

Abstract

Proton or lithium diffusion is a critical electrode process that occurs in manganese dioxide electrode materials during cycling of either aqueous (alkaline) or non-aqueous (lithium) batteries. The structural and electrochemical properties of a number of hydrated alpha-phase manganese dioxide compounds ($\alpha\text{-MnO}_2\bullet n\text{H}_2\text{O}$; $n\approx 0.2\text{-}0.33$), the heat-treated products ($n\approx 0\text{-}0.1$), as well as their more stable lithia-doped derivatives, $\alpha\text{-}[x\text{Li}_2\text{O}]\bullet\text{MnO}_2$ ($0\leq x\leq 0.25$), have been investigated. Inelastic neutron scattering was used as a means to differentiate and interrogate the key proton or water interactions in these MnO_2 structures. The neutron spectra reveal excitations associated with the hydrogen vibrations centered at 4, 13, 16, 26, 36, 65, 90, 110, 140, 200, 420, and 490 meV in the alpha-phase and 14, 24, 65, 84, 100, 202, and 425 meV in the lithia-doped derivatives. Band assignments were carried out by comparing these frequencies with those reported for structurally similar hydrated $\gamma\text{-MnO}_2$ compounds and by comparison with infrared data.

Keywords: manganese dioxide, lithium battery, neutron scattering

Correspondence:

Dr. Christopher S. Johnson
CMT, Building 205
Argonne National Laboratory
Argonne, IL 60439-4814

FAX: (630) 252-4167
Email: johnsoncs@cmt.anl.gov

The submitted manuscript has been created by the University of Chicago as Operator of Argonne National Laboratory ("Argonne") under Contract No. W-31-109-ENG-38 with the U.S. Department of Energy. The U.S. Government retains for itself, and others acting on its behalf, a paid-up, nonexclusive, irrevocable worldwide license in said article to reproduce, prepare derivative works, distribute copies to the public, and perform publicly and display publicly, by or on behalf of the Government.

MASTER

DISTRIBUTION OF THIS DOCUMENT IS UNLIMITED

DISCLAIMER

This report was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.

DISCLAIMER

**Portions of this document may be illegible
in electronic image products. Images are
produced from the best available original
document.**

1. Introduction

Manganese dioxide (MnO_2) materials are increasingly being chosen as the active electrode material in non-aqueous, primary lithium (Li/non-aqueous organic electrolyte/ MnO_2), and secondary (rechargeable) lithium-ion batteries (Li_xC_6 /non-aqueous organic electrolyte/ $\text{Li}_{1-x}\text{Mn}_2\text{O}_4$) [1]. MnO_2 is attractive because of its relatively low cost compared to either nickel in Ni-MH (metal-hydride) or cobalt in lithium-ion (LiCoO_2) secondary batteries. Moreover, recent data has demonstrated good performance from MnO_2 -based chemistries as well, with cells exhibiting high energy/power density properties [2].

The development of non-aqueous MnO_2 cells has evolved as a result of the use of γ - MnO_2 initially as a primary cathode in aqueous alkaline batteries (Zn/KOH/ MnO_2). In the aqueous cell, the discharge process is governed by the diffusion and reaction of protons (H^+) on the surface of the γ - MnO_2 electrode particle. Due to the chemical similarity of Li^+ to H^+ , it is reasonable to expect that the behavior of the lithium cation in a non-aqueous cell may be equivalent. In fact, specially heat-treated γ - MnO_2 functions quite well, giving 3 volts in a lithium non-aqueous cell and yielding capacities in the range of 200 mAh/g, equivalent to the insertion of about 0.65 Li per mole of MnO_2 [3]. However, these particular materials have limited reversibility which hinders their rechargeability.

Our studies have focused on the preparation and characterization of a class of new alpha-phase (α - MnO_2) materials which possess good rechargeability in terms of capacity and rate in 3 V lithium cells. The objective is to obtain information about the role hydrogen plays in H-bonding and diffusion in these materials. In turn, data from inelastic neutron scattering studies may help us formulate a good comparison between hydrogen and lithium and allow us a better understanding of lithium/ MnO_2 batteries. Due to the adverse reaction with metallic lithium or a lithiated negative electrode, the minimization of water in a non-aqueous lithium cell is necessary for its proper operation.

2. Experimental

Hydrated alpha manganese dioxide ($\alpha\text{-MnO}_2\bullet\text{H}_2\text{O}$; $n\approx0.2\text{-}0.33$), its heat-treated product and the subsequent lithium-oxide stabilized product ($\alpha\text{-}[\text{xLi}_2\text{O}]\$Mn\text{O}_2$) were synthesized as previously described [4]. A detailed description of the electrochemical experiments may be found in reference 4.

Inelastic neutron scattering experiments on two hydrated samples ($\alpha\text{-MnO}_2\bullet0.25\text{H}_2\text{O}$, $\alpha\text{-MnO}_2\bullet0.33\text{H}_2\text{O}$), a dehydrated ($\alpha\text{-MnO}_2$) and partially dehydrated product ($\alpha\text{-MnO}_2\bullet0.10\text{H}_2\text{O}$), a subsequent lithium-oxide stabilized ($\alpha\text{-}[0.143\text{Li}_2\text{O}]\bullet\text{MnO}_2$) product, and a similar deuterated sample ($\alpha\text{-MnO}_2\bullet0.12\text{D}_2\text{O}$), were carried out using the HRMECS spectrometer at IPNS. Incident energies of 50, 150, 170, 250, 350, and 600 meV were chosen and the sample temperature was controlled in a range of 6-15 K.

3. Results

Alpha- MnO_2 has a framework structure containing MnO_6 octahedra. It has been commonly believed that the $\alpha\text{-MnO}_2$ structure is stabilized by a foreign cation such as K^+ , NH_4^+ , or Ba^{2+} within the 2×2 tunnel interstitial space of the framework [5]. From our synthesis method, we have succeeded to synthesize an $\alpha\text{-MnO}_2$ structure devoid of such "stabilizing" cations [6]. The process yields a product which is highly crystalline and dense and which has a tetragonal symmetry ($I4/m$). The average unit cell parameters, determined from powder X-ray diffraction on 10 separate hydrated samples, are $a = b = 9.813$ Å and $c = 2.850$ Å. Analyses of neutron diffraction data by Rietveld profile refinement have shown unequivocally that molecular water (H_2O , or possibly H_3O^+) is present within the 2×2 tunnels of the structure and the oxygen atom of the water is located at the center of the tunnel in the (0.0,0.0,0.5) unit cell position [4].

The hydrated $\alpha\text{-MnO}_2$ can be heat-treated at 275 °C in air to remove surface-bound water and water within the 2×2 tunnels. Thermogravimetric data collected on this material showed that about 4-5 wt.-% water was present as structural (2×2) tunnel water,

and 2-3 wt.-% was present as surface (interfacial) water. Furthermore, the α -MnO₂ structure itself remains intact at temperatures up to 400°C. Structure analysis of the heat-treated samples confirmed earlier reports that the MnO₂ framework structure remains intact during dehydration [7] and showed that, on average, the unit cell contracts by about 0.5% ($a = b = 9.763 \text{ \AA}$, $c = 2.860 \text{ \AA}$).

The α -MnO₂ framework structure can be stabilized by an exchange of the H₂O component in the α -MnO₂•nH₂O hydrated phase with Li₂O, and that n can reach a value of 0.15 (in nLi₂O) without structural degradation of the α -MnO₂ framework [4]. The unit cell parameters of several samples with varying Li:Mn mole ratios were calculated from powder X-ray diffraction data. The unit cell expands, almost linearly, with increasing lithium content to a Li : Mn ratio of 0.3 : 1 without the appearance of a second phase; with this ratio the unit cell parameters are $a = b = 9.932 \text{ \AA}$, and $c = 2.852 \text{ \AA}$. Chemical analysis of lithium-oxide doped products quantitatively verified the reactant Li : Mn mole ratio. The O²⁻ anions occupy crystallographic positions within the 2 x 2 tunnels (Fig. 1); the lithium cations appear to be coordinated octahedrally to the oxygen atoms of the framework structure and those located within the 2 x 2 tunnels [4].

Interactions between the H₂O molecules with the MnO₂ structure are important to address because of their structural equivalency to the Li₂O molecule. These were assessed by inelastic neutron scattering measurements. The inelastic neutron cross-section scattering coefficient of hydrogen is at least one order of magnitude greater than that of manganese or oxygen. Thus, the intensities of bands in the energy transfer spectra are predominantly due to hydrogen interactions within the structure. Fig. 2 shows a representative, baseline-corrected spectrum of α -MnO₂•0.25H₂O obtained with a neutron incident energy of 600 meV. Fig. 2 also displays the spectrum of the same sample after heat treatment at 275 °C. This spectrum is essentially featureless in terms of hydrogen interactions and confirms that water is removed from α -MnO₂ on heat treatment. Note that the presence of a large continuum background in the spectrum of the hydrated sample

suggests significant mobility of hydrogen ions. These spectra are similar to a previously reported neutron scattering survey of other manganese oxide structures which have been used in rechargeable lithium batteries [8]. A detailed analysis of the neutron results in conjunction with the electrochemical properties of the Li-inserted MnO_2 system will be given elsewhere.

Acknowledgments

We gratefully acknowledge support for this work from the U. S. Department of Energy's Advanced Battery Research Program, Chemical Sciences Division, Office of Basic Energy Science. Work performed at Argonne is supported by the U.S.DOE-BES under Contract No.W-31-109-ENG-38.

References

- [1] M. M. Thackeray, M. H. Rossouw, A. de Kock, A. P. de la Harpe, R. J. Gummow, K. Pearce, and D. C. Liles, *J. Power Sources*, 43-44 (1993), 289.
- [2] G. Pistoia, and A. Antonini, *J. Electrochem. Soc.*, 144 (1997) 1553.
- [3] T. Nohma, Y. Yamamoto, I. Nakane, and N. Furakawa, *J. Power Sources*, 39 (1992) 51.
- [4] C. S. Johnson, D. W. Dees, M. F. Mansuetto, M. M. Thackeray, D. R. Vissers, D. Argyriou, C.-K. Loong, and L. Christensen, *J. Power Sources*, in press (1997).
- [5] A. F. Wells, *Structural Inorganic Chemistry* (4th edition), 458 (1975), Clarendon Press, Oxford, UK.
- [6] M.H. Rossouw, D.C. Liles, and M.M. Thackeray, *Prog. Batt. and Batt. Mat.* 15 (1996) 8.
- [7] M.H. Rossouw, D.C. Liles, M. M. Thackeray, W.I.F. David, and S. Hull, *Mat. Res. Bull.* 27 (1992) 221.
- [8] C. Cachet, A. Belushkin, I. Natkaniec, A. Lecerf, F. Filliaux, and L. T. Yu, *Physica B* 213/214 (1995) 827.

Figure Caption:

Fig. 1 The positions of the oxygen atoms within the 2 x 2 tunnels of the lithiated α -MnO₂ structure.

Fig. 2 Representative base-line corrected inelastic neutron scattering spectra for α -MnO₂•0.25H₂O (open circles) and the same sample after heat treatment at 275 °C (filled circles).

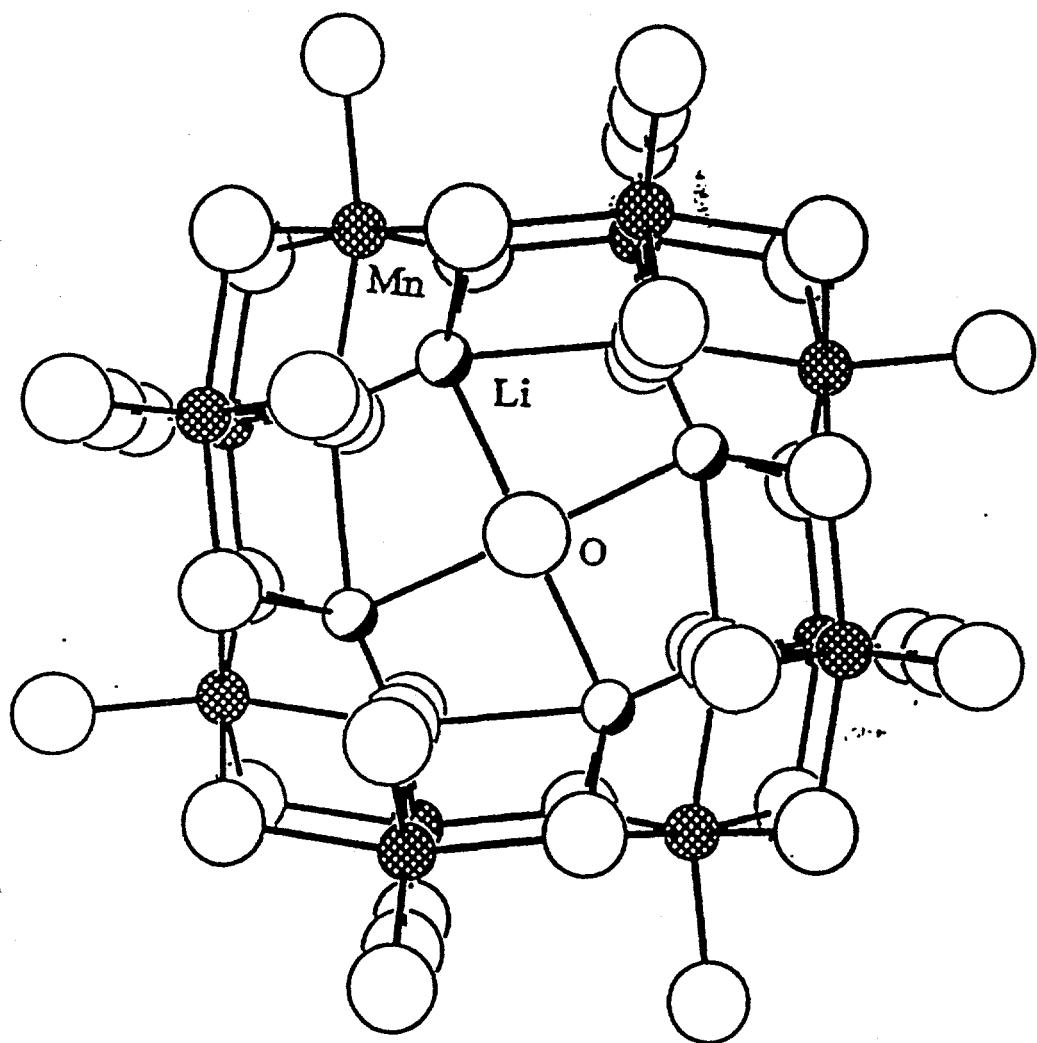


Fig. 1 "Spectroscopic Study of ... " C. S. Johnson *et. al.*

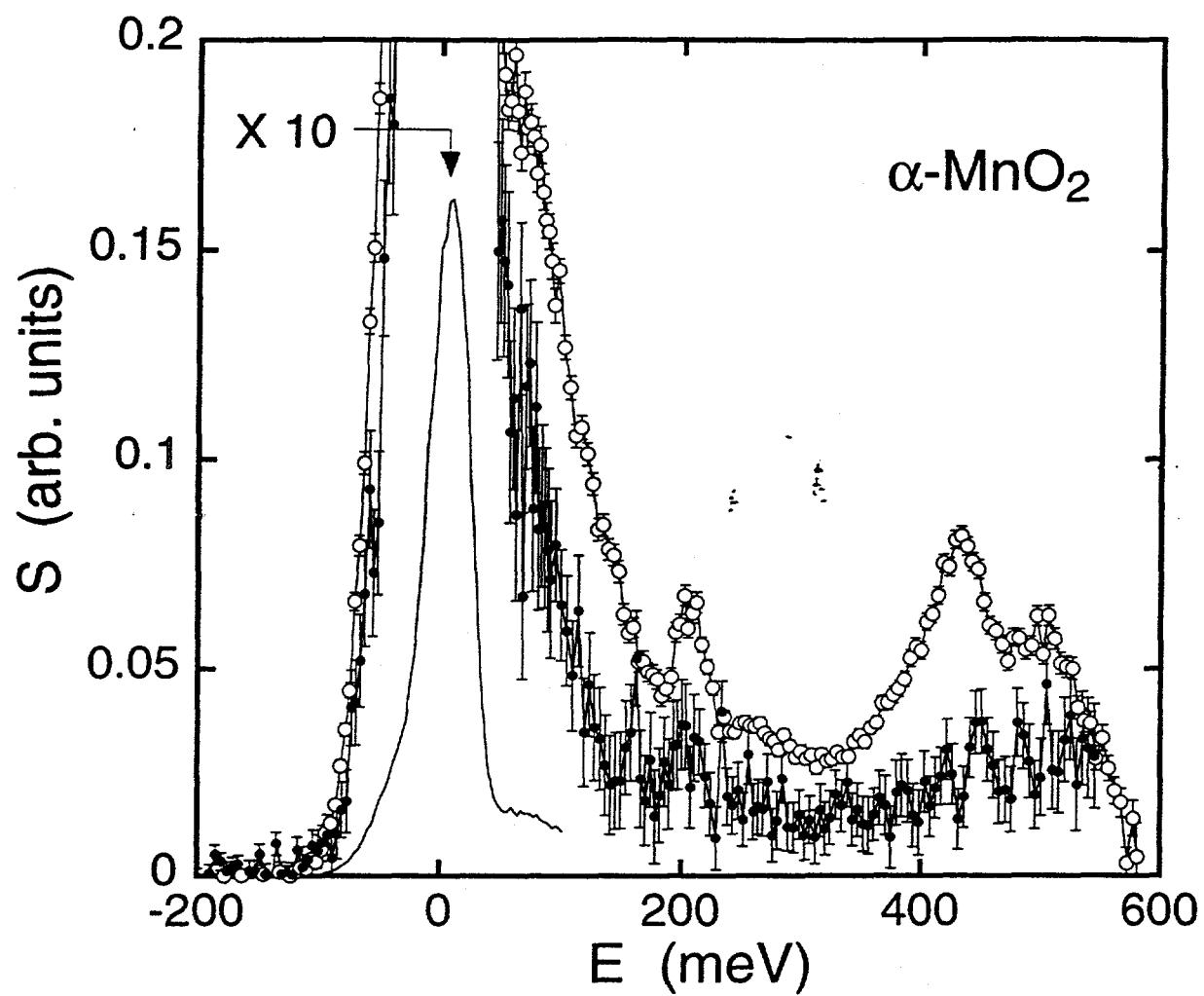


Fig. 2: "Spectroscopic Study of ... " C. S. Johnson et. al.