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PREDICTING AND EVALUATING NEW WATER SPLITTING MATERIALS FOR SOLAR THERMOCHEMICAL HYDROGEN PRODUCTION

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24th International Conference on Solid State Ionics, London, UK

3C2 Catalysis

11:15 am – 11:45 am, Wednesday, July 17, 2024 (Room Fleming) - Keynote

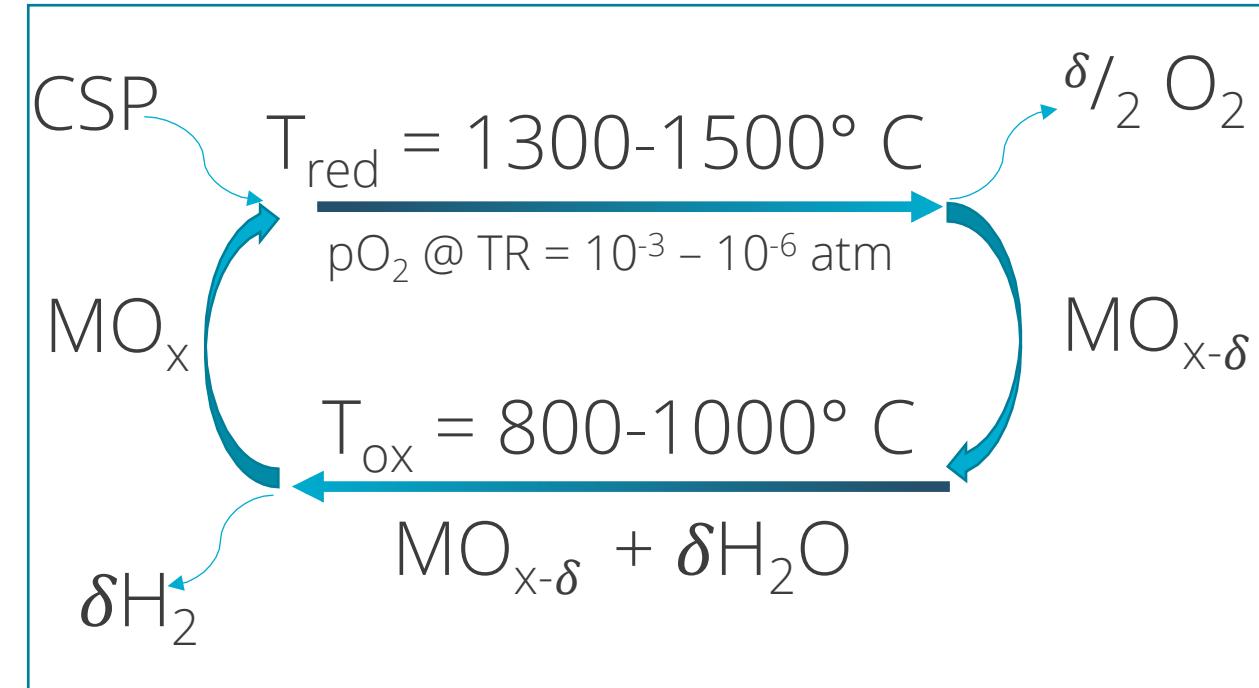
Outline

- Solar thermo-chemical hydrogen production (STCH) background and the US DOE HydroGEN consortium
- Addressing needs for STCH community
 - Identification of benchmarking metrics and applying them to exemplar materials
 - New materials identification aided by computational discovery with machine learning approach
 - Revisiting oxygen non-stoichiometry in $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_{3-\delta}$ at STCH reducing temperatures (>1200 °C), and new High Entropy Perovskite Oxide (HEPO)

Solar Thermo-Chemical Hydrogen Production (STCH)

Sunlight (heat) + water → Hydrogen

National Solar Thermal Test Facility at Sandia National Labs

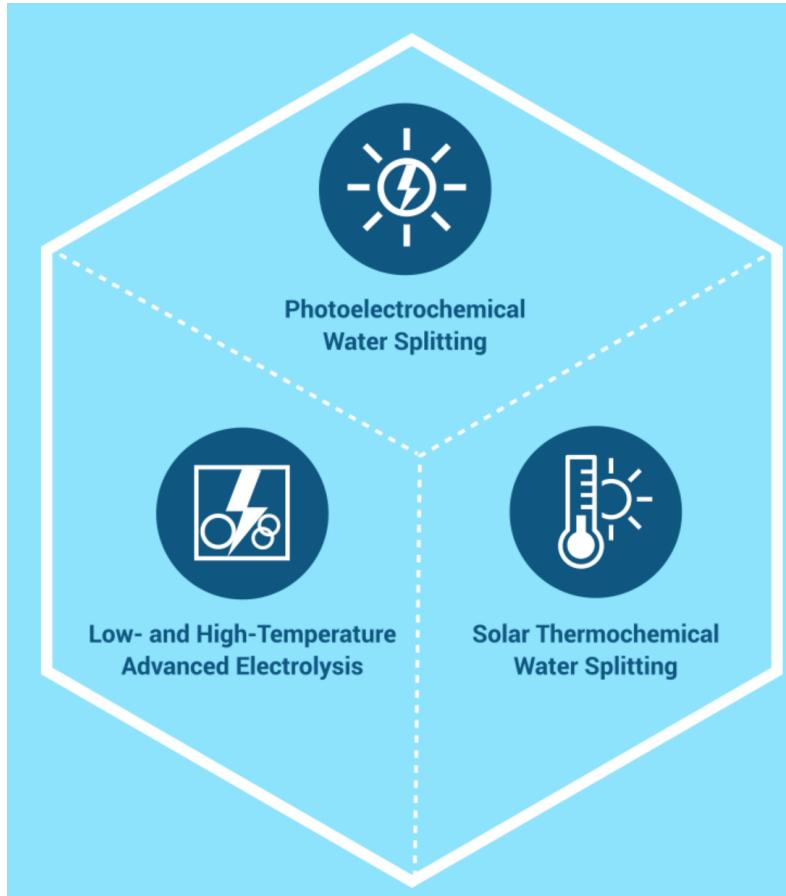


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- Single phase materials to avoid cycling degradation
- Oxides that “breathe” oxygen needed → non-stoichiometric oxides
- Goldilocks reduction enthalpy (H_r)
- Also use heat from other sources → *Thermo-chemical hydrogen production (TCH)*

Department of Energy HydroGEN Consortium (H2AWSM.org)

Research focus areas



Leverage capabilities at national laboratories

- National Renewable Energy Laboratory (Lead)
- Lawrence Berkeley National Laboratory
- Sandia National Laboratories
- Idaho National Laboratory
- Lawrence Livermore National Laboratory

Collaboration with industry and academia seedlings supported by HydroGEN funding opportunities

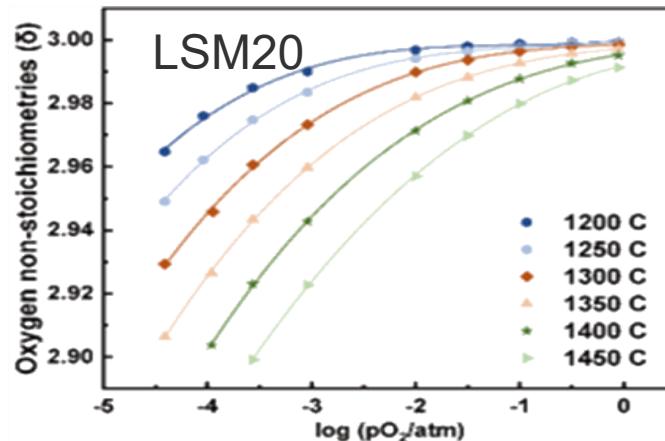
Sandia node capabilities used for this presentation:

- Thermo-gravimetric analysis (Sean Bishop)
- STCH reactor (Tony McDaniel)
- Electron microscopy and composition analysis (Josh Sugar)

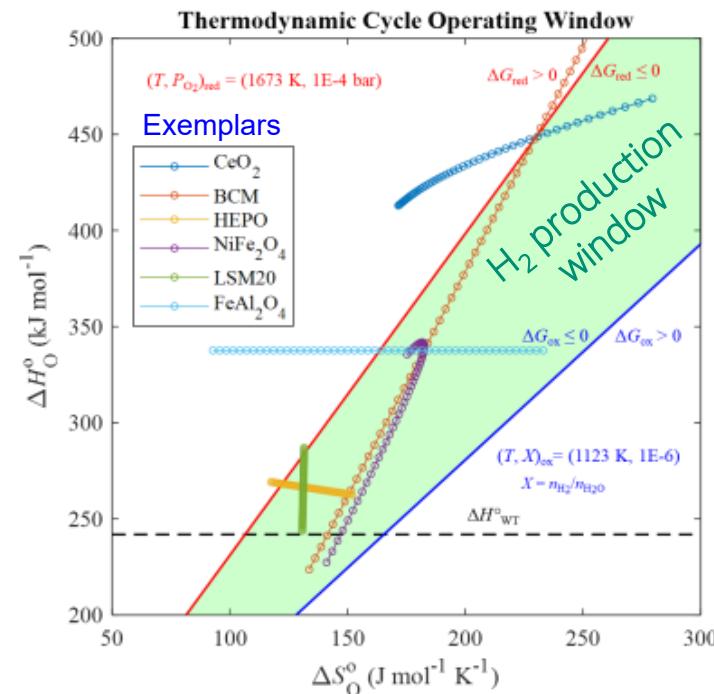
Exemplar Material Viability Study

- Define STCH metrics
- Benchmark exemplar materials against state of the art (CeO_2)
- Identify technology gaps

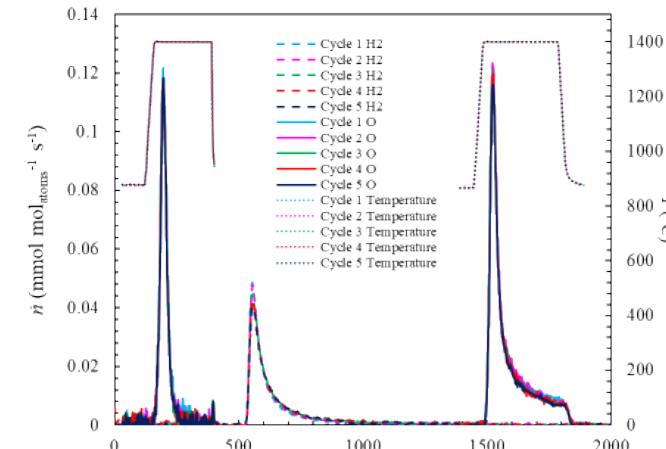
Thermodynamic parameters (from thermo-gravimetric analysis)



BCM: $\text{BaCe}_{0.25}\text{Mn}_{0.75}\text{O}_3$
 HEPO: $\text{La}_{1/6}\text{Pr}_{1/6}\text{Nd}_{1/6}\text{Gd}_{1/6}\text{Ba}_{1/6}\text{Sr}_{1/6}\text{MnO}_3$
 LSM20: $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_3$



Hydrogen production and kinetic parameter (from flow reactor)



Cycle efficiency estimation

$$\lambda = \frac{1}{2\Delta p^*} = \frac{\dot{n}_{\text{N}_2}}{2\Delta \dot{n}_{\text{O}_2}} = \frac{\dot{n}_{\text{N}_2}}{\Delta \dot{n}_{\text{H}_2}}$$

$$p^* = \frac{p_{\text{O}_2}}{p_{\text{sys}} - p_{\text{O}_2}} = \frac{\dot{n}_{\text{O}_2}}{\dot{n}_{\text{inert}}}$$

$$\theta = \frac{\dot{n}_{\text{H}_2}}{\dot{n}_{\text{H}_2\text{O}}}$$



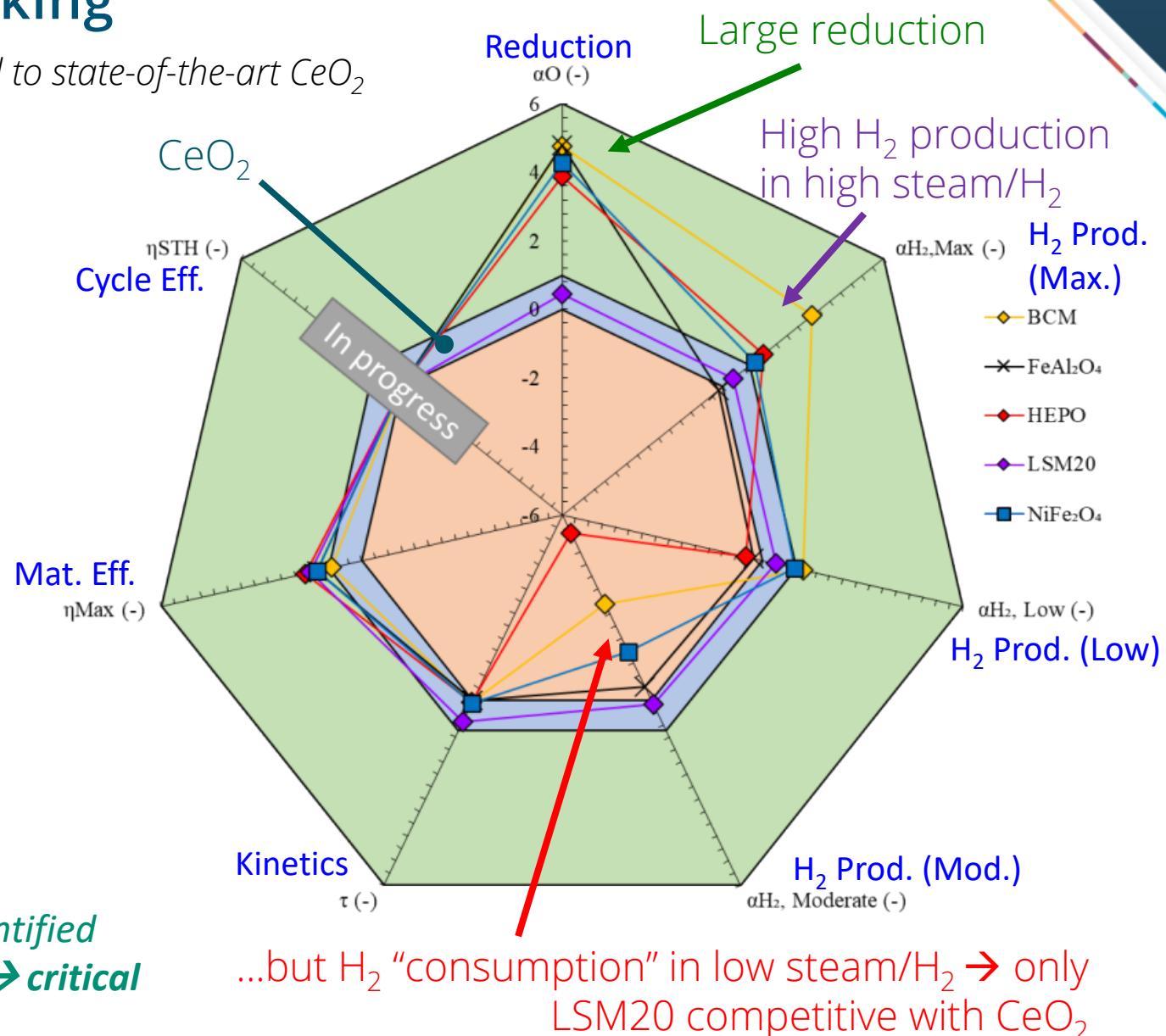
Metrics and Exemplar Benchmarking

Exemplars normalized to state-of-the-art CeO_2

Metrics

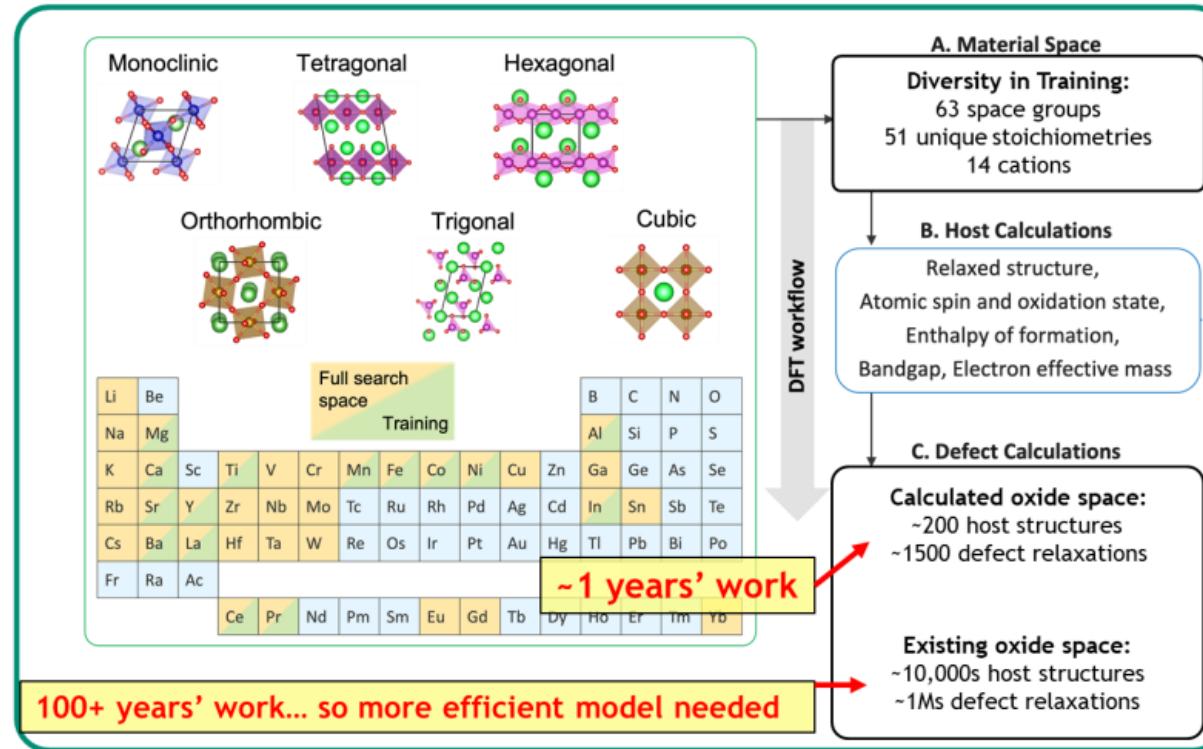
Metrics	Descriptor	Target Values
Cycle Efficiency (STH)	Solar-to-hydrogen conversion efficiency derived from detailed cycle analysis using a thermodynamic model based on specific plant operational assumptions	$\eta_{\text{STH}} > 26\%$
Material Efficiency	$\frac{\Delta G_{\text{WS}}^0}{\Delta H_{\text{O}}^0}$ is the maximum possible thermal efficiency of the two-step process. (ΔG_{WS}^0 evaluated at 25 °C)	$\eta_{\text{Max}} > 50\%$
Reduction Capacity	mmol O / mol atom in solid reduced @ neutral low condition	$\alpha_{\text{O}} > 5$
STCH Capacity (Maximum Yield)	mmol H ₂ / mol atom in solid reduced @ neutral low condition, oxidized in pure H ₂ O @ optimal T _{ox} for material	$\alpha_{\text{H}_2, \text{Max}} > 5$
STCH Capacity (Low Yield)	mmol H ₂ / mol atom in solid reduced @ neutral low condition, oxidized in steam-to-fuel ratio H ₂ O/H ₂ = 1000 @ optimal T _{ox} for material	$\alpha_{\text{H}_2, \text{Low}} > 2.5$
STCH Capacity (Moderate Yield)	mmol H ₂ / mol atom in solid reduced @ neutral low condition, oxidized in steam-to-fuel ratio H ₂ O/H ₂ = 100 @ optimal T _{ox} for material	$\alpha_{\text{H}_2, \text{Mod}} > 1$
Kinetic Performance	Time to 90% of $\alpha_{\text{H}_2, \text{Max}}$ in pure H ₂ O at optimal T _{ox} for specific material in a dispersed powder configuration	$\tau > 0.20$

- *Evaluation framework created and metrics identified*
- *Weakness of exemplars in low steam/H₂ ratio → critical need for new materials*

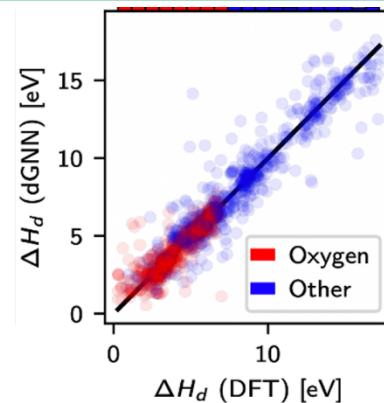


Discovery of New STCH Materials Aided by Machine Learning

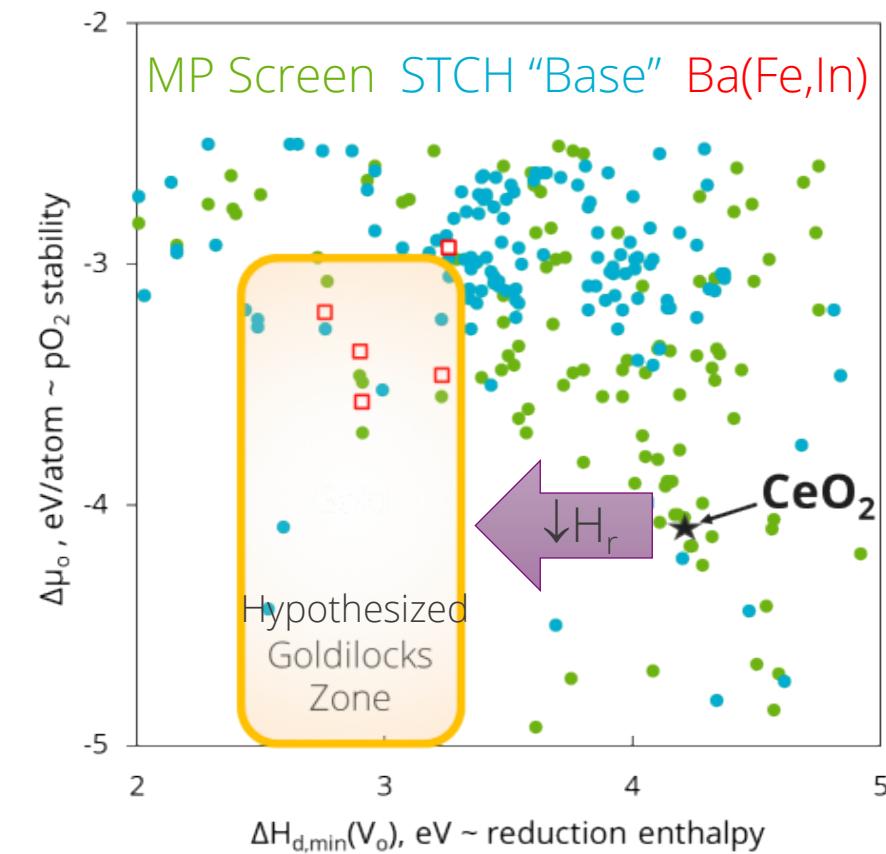
First-principles DFT workflow is robust but costly (using NRELMatDb hosts)



ML screens 10,000's of MP structures in minutes that would take 1,000's of DFT months



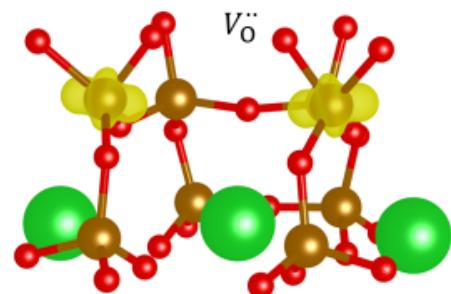
Expected $\Delta H_{0,d}$ MAE for unseen compounds < 450 meV (threshold for ML to be predictive).



→ Examples: Identified $\text{Ba}(\text{Fe,In})_2\text{O}_4$ and $(\text{Ba},\text{Sr})_6(\text{La},\text{Y})_2\text{Fe}_4\text{O}_{15}$

BaFe_2O_4 : $\text{V}_\text{O}^{\cdot\cdot}$ Point Defect Mechanism, and Al Addition to Increase Temperature Stability

BaFe_2O_4
Orthorhombic (Bb21m) phase



DFT \rightarrow oxygen vacancy preferred vs. cation defects

Melting of BFO in low pO_2 mitigated by Al substitution!

1400 °C in Ar (~20 ppm O_2)



$\text{BaAl}_{0.4}\text{Fe}_{1.6}\text{O}_4$ – Hexagonal ($\text{P}6_3$) phase

→ Enables higher reduction temperature and resistance to densification

Higher melting point expected with Al

- BaAl_2O_4 (hexagonal $\text{P}6_3$, stuffed tridymite) melting point ~ 1820 °C
- MgAl_2O_4 (cubic $\text{Fd}-3\text{m}$, spinel) melting point ~ 2130 °C

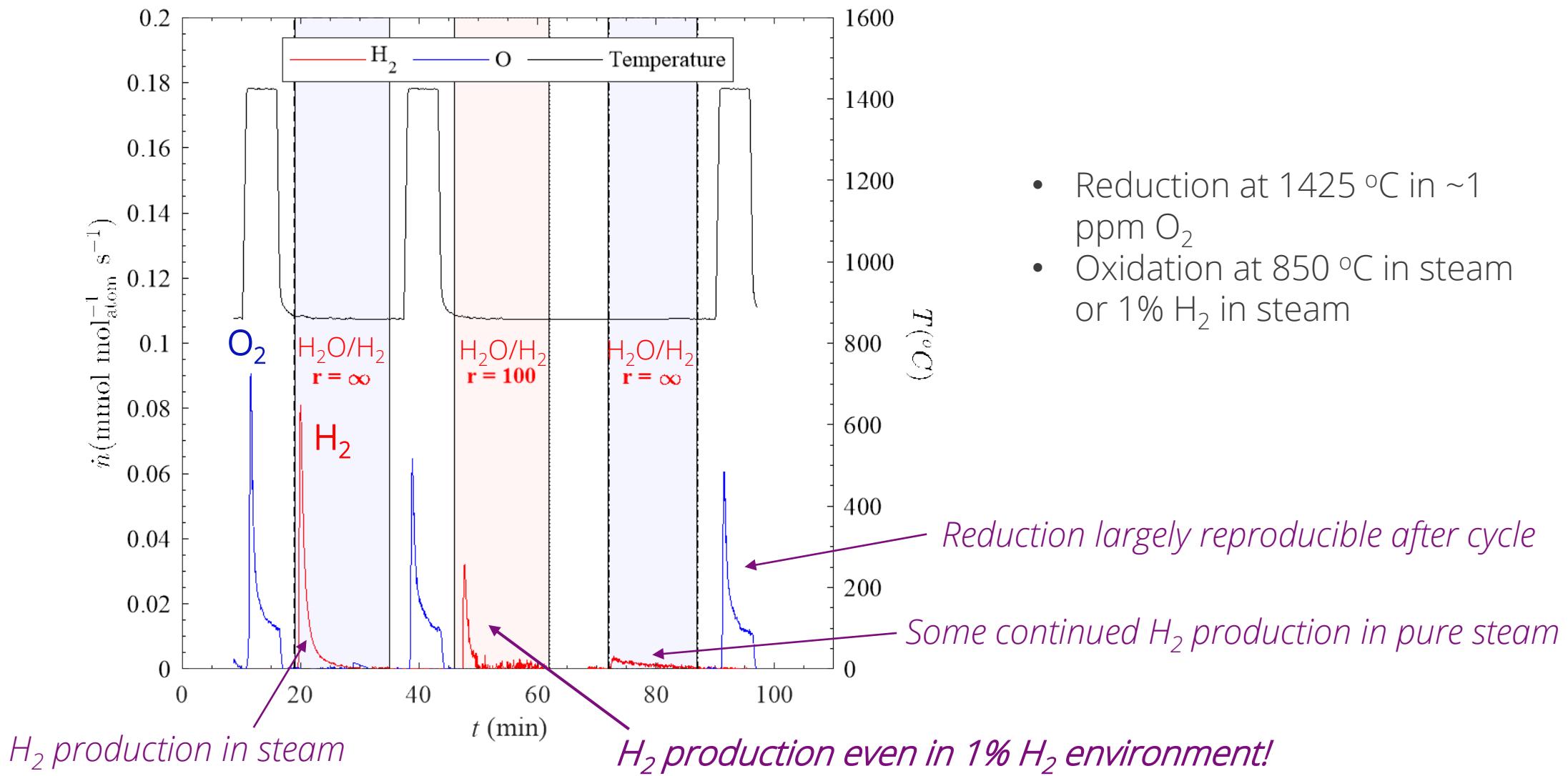
$\text{BaAl}_{0.4}\text{Fe}_{1.6}\text{O}_4$ stable at 1450 °C in air!

BaFe_2O_4 $T_{melt,air} \sim 1420$ °C

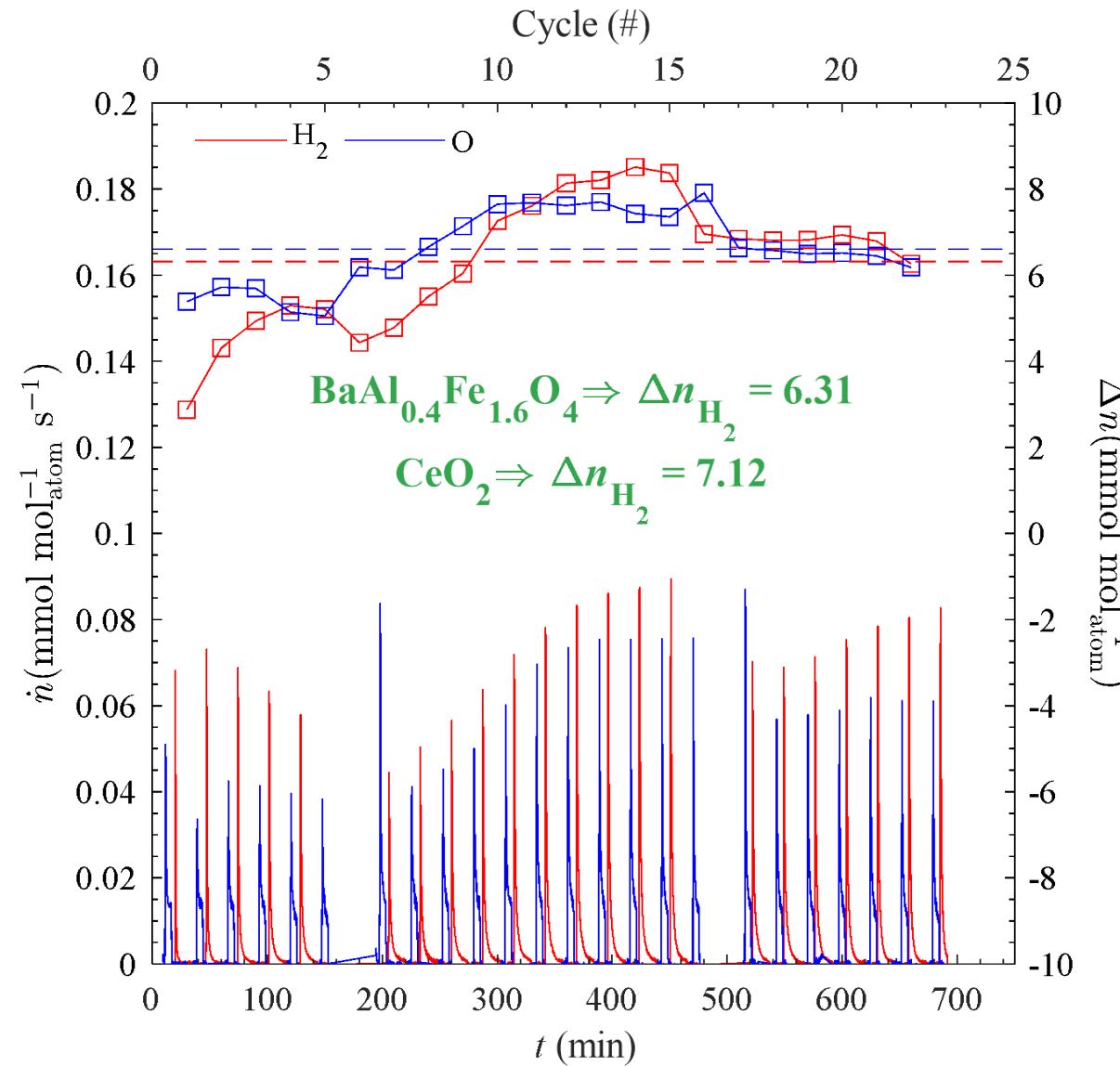


$\text{BaAl}_{0.4}\text{Fe}_{1.6}\text{O}_4$: Water Splitter in Low Steam/ H_2 !

H_2 production measurement using flow reactor



H₂ Production Over 20 Cycles



- Reduction at 1425 °C in ~1 ppm O₂
- Oxidation at 850 °C in steam

Comparable H₂ production performance to CeO₂ retained for 20 cycles!

Severe Change in Morphology and Phase After Cycling

Initial powder



After 20 redox cycles

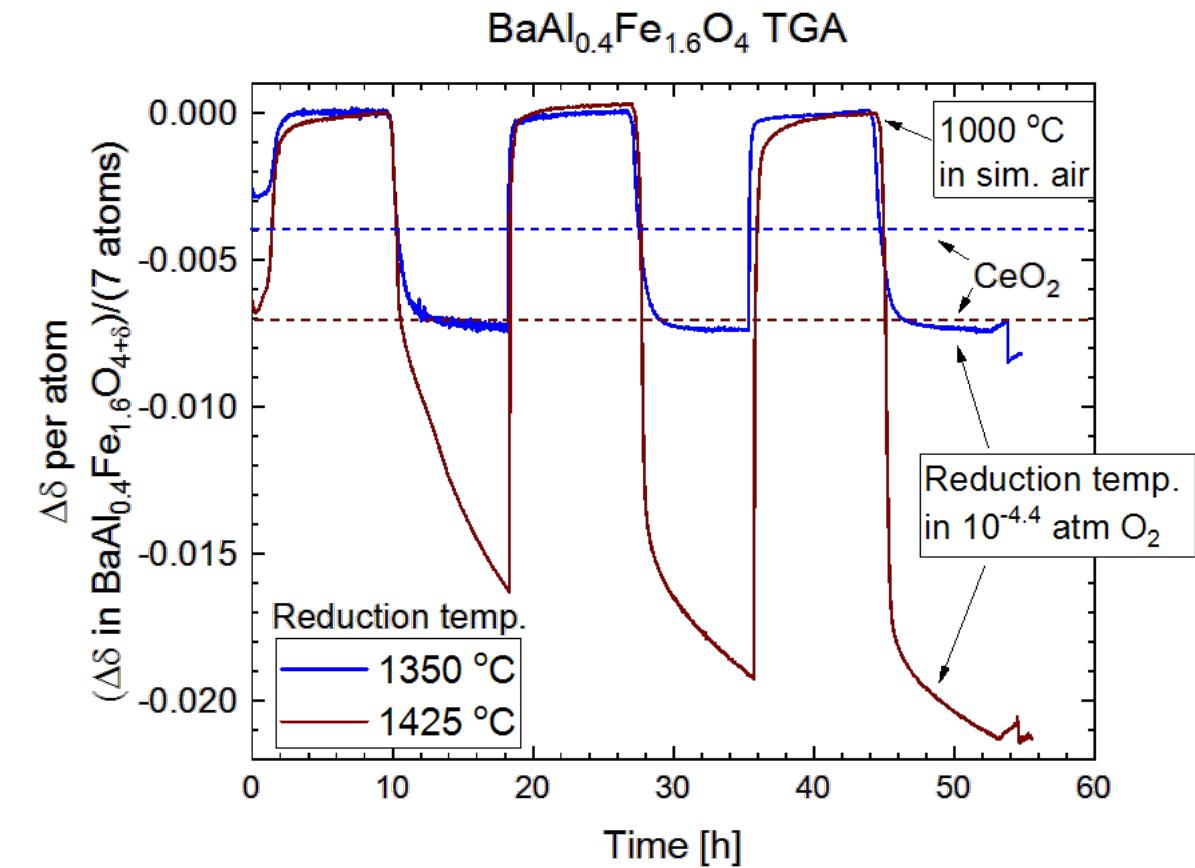


Post-cycled sample initial diffraction analysis at SLAC

- BaAl_2O_4 ($\text{P}6_3$) is 20.34 wt% → starting phase
- BaFe_2O_4 ($\text{Cmc}21$) is 75.17 wt% }
- BaFeO_3 ($\text{P}6_3/\text{mmc}$) is 4.48 wt% }

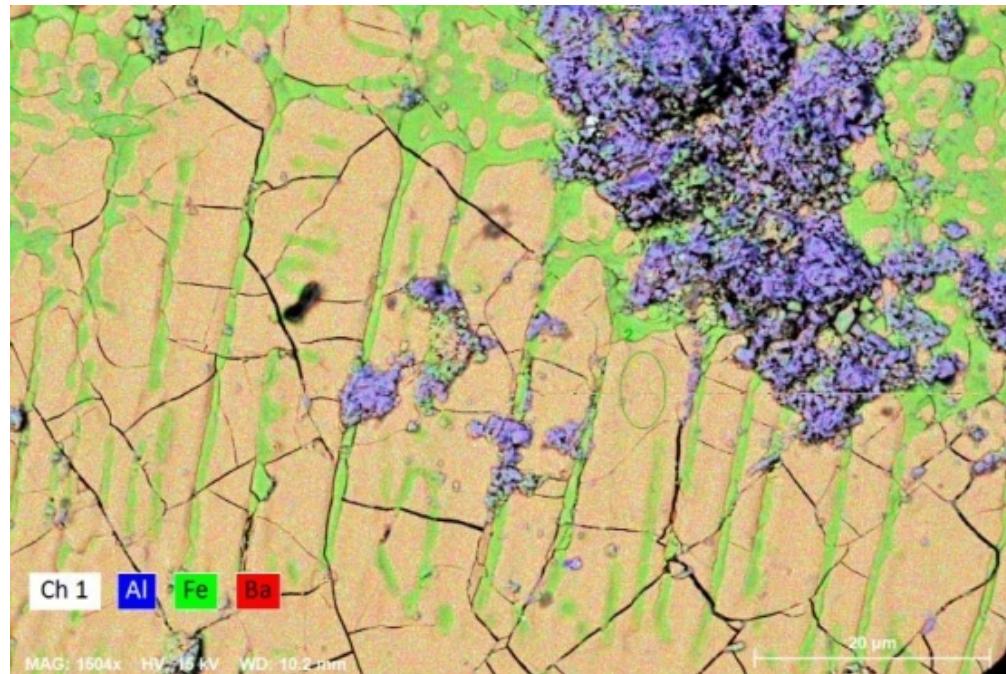
Decomposition products

Long-term reduction at 1425 °C
→ continuous oxygen release



Detailed Compositional Analysis – 3-phases

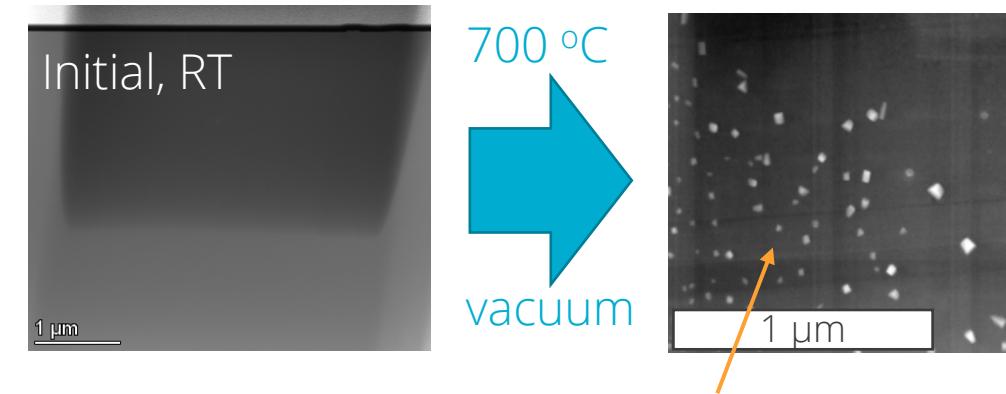
EDS map of post-20 cycled sample



- Purple phase consistent with $\sim\text{BaAlFeO}_4$ (the expected $\text{P}6_3$ structure)
- Tan phase consistent with $\sim\text{BaFeO}_3$, with possible melting
- Green phase consistent with $\sim\text{BaFe}_2\text{O}_4$, with possible melting

In situ reduction in TEM → evolution of decomposition

FIB cross-section of $\text{BaAl}_{0.4}\text{Fe}_{1.6}\text{O}_4$



Fe-rich particles formed

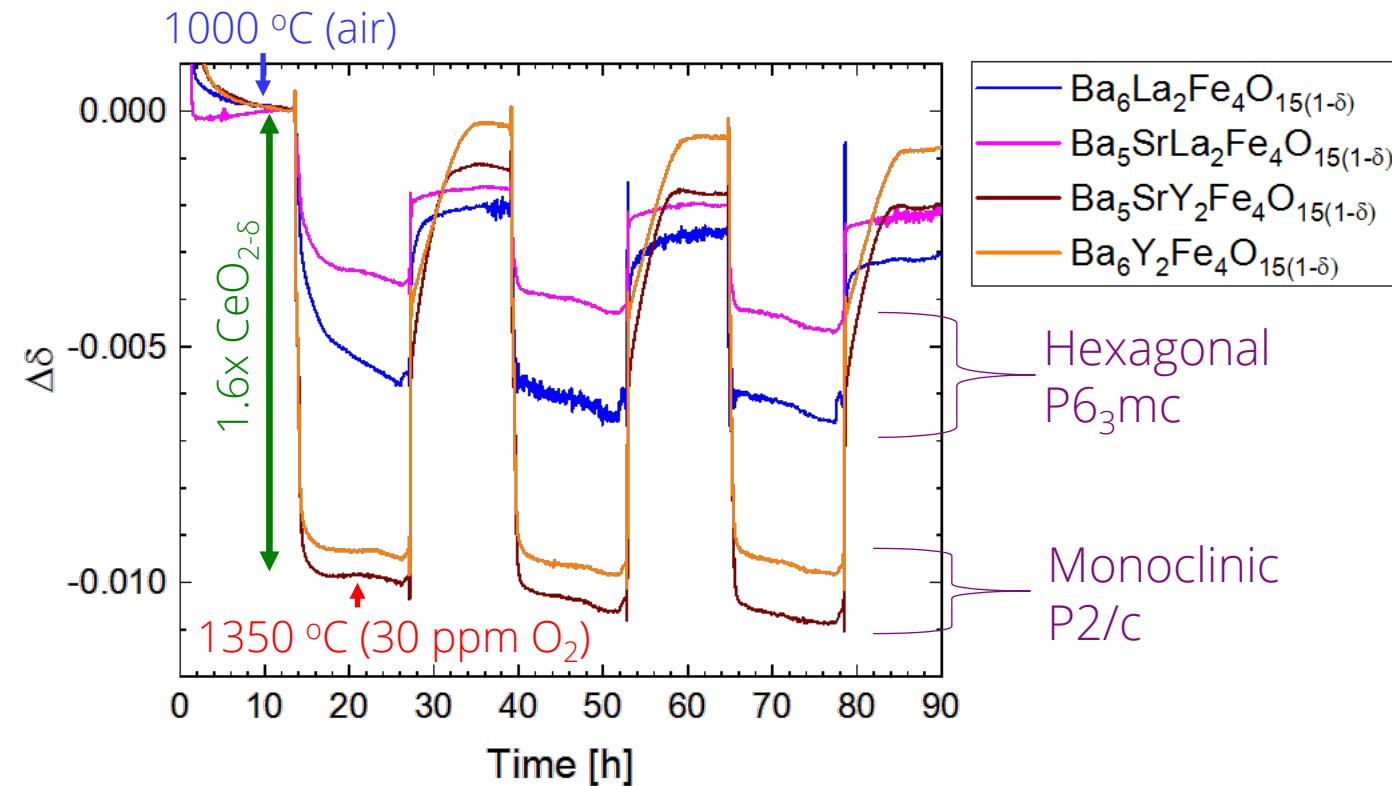
Future Work:

- investigating role of each phase in water splitting and durability
- HTXRD in simulated water splitting conditions

$(\text{Ba},\text{Sr})_6\text{La}_2\text{Fe}_4\text{O}_{15}$ and $(\text{Ba},\text{Sr})_6\text{Y}_2\text{Fe}_4\text{O}_{15}$ ML Predicted Water Splitters



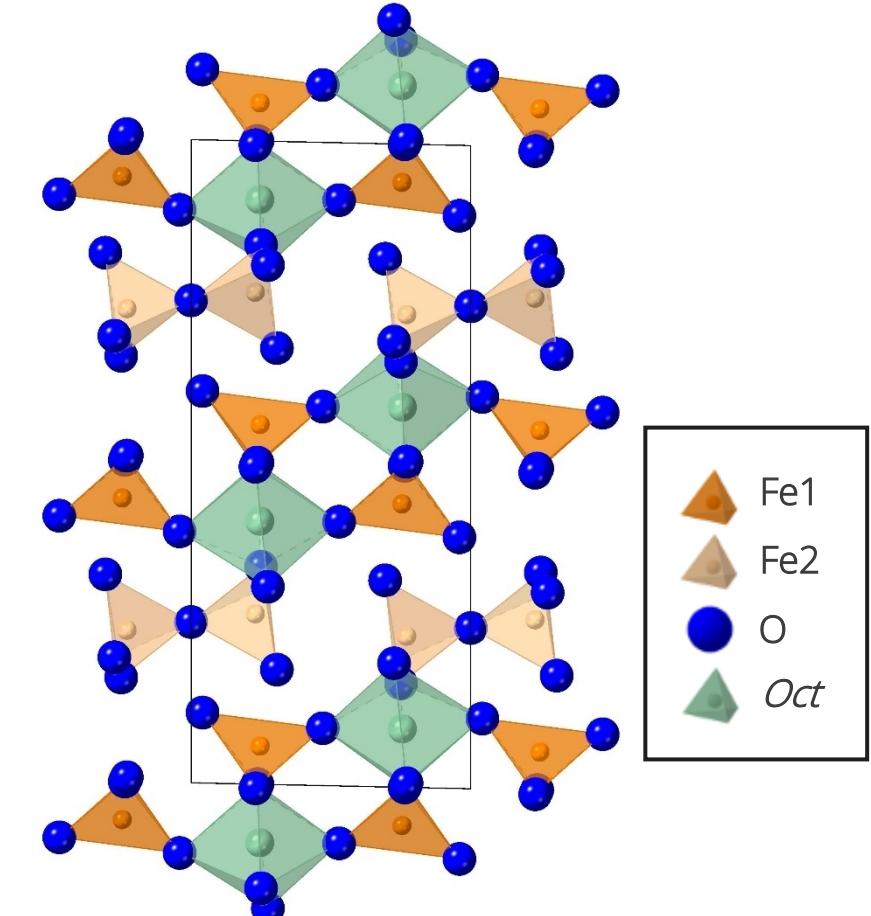
Generalized monoclinic structure (excluding Ba,Sr)
Fe apex sharing dimers (not in hexagonal structure)



STCH screening protocol¹ → $\text{Ba}_6\text{Y}_2\text{Fe}_4\text{O}_{15}$ is best

¹Sanders *et al.*, *Front. Energy Res.* 10:856943 (2022)

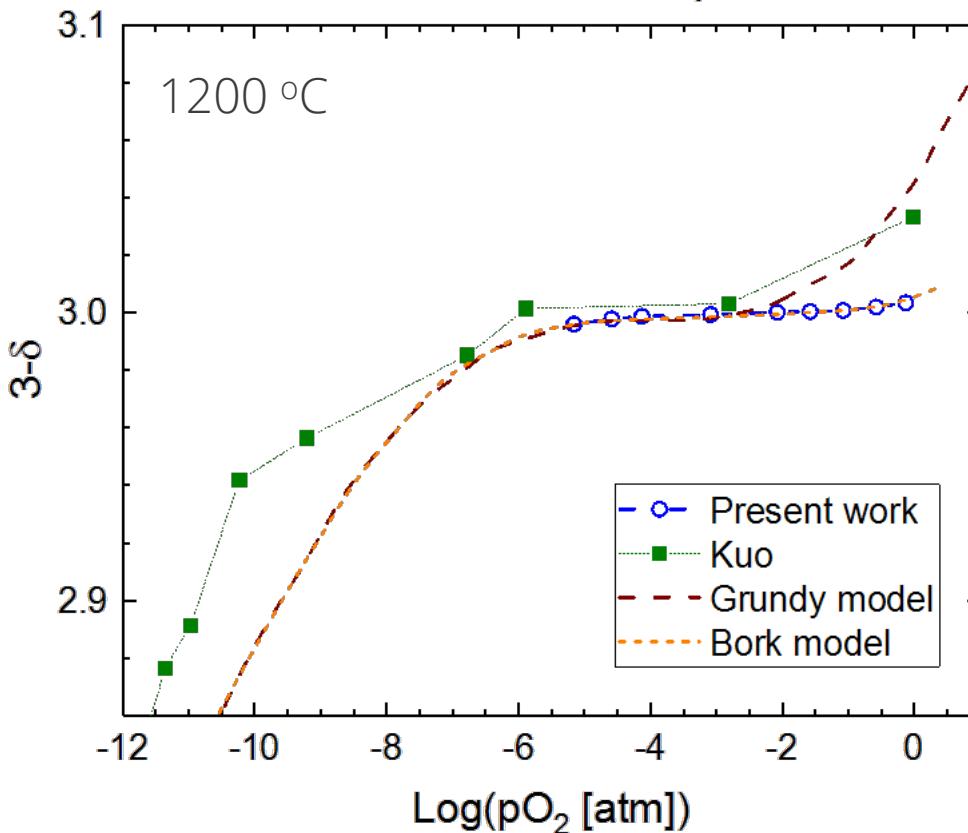
Monoclinic → $\Delta\delta > \text{CeO}_2$



Future Work → Flow reactor measurements

Re-Visiting $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_3$: Oxygen Stoichiometry at High Temp.

Limited δ data above 1200 °C → evaluate at higher temperatures (e.g., STCH conditions)



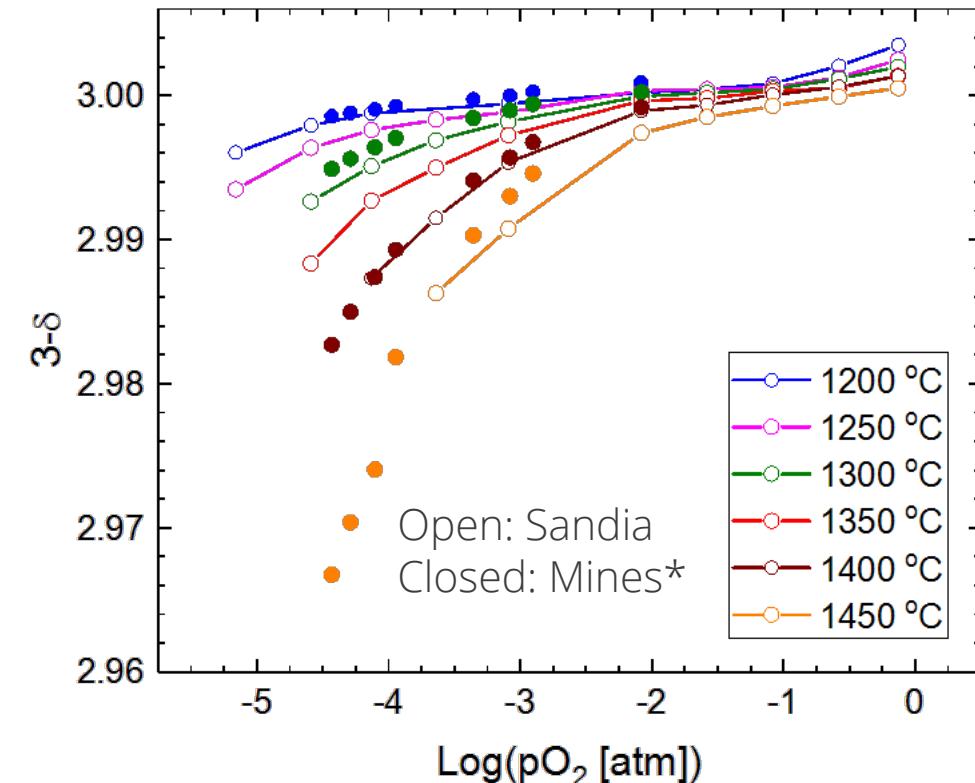
Less super-stoichiometry than previously estimated, in agreement with Bork model

Kuo et al., J. Sol. State Chem., 83 (1989) 52

Bork et al., Adv. Energy Mater. 7 (2017) 1601086

Grundy et al., CALPHAD 28 (2004) 191

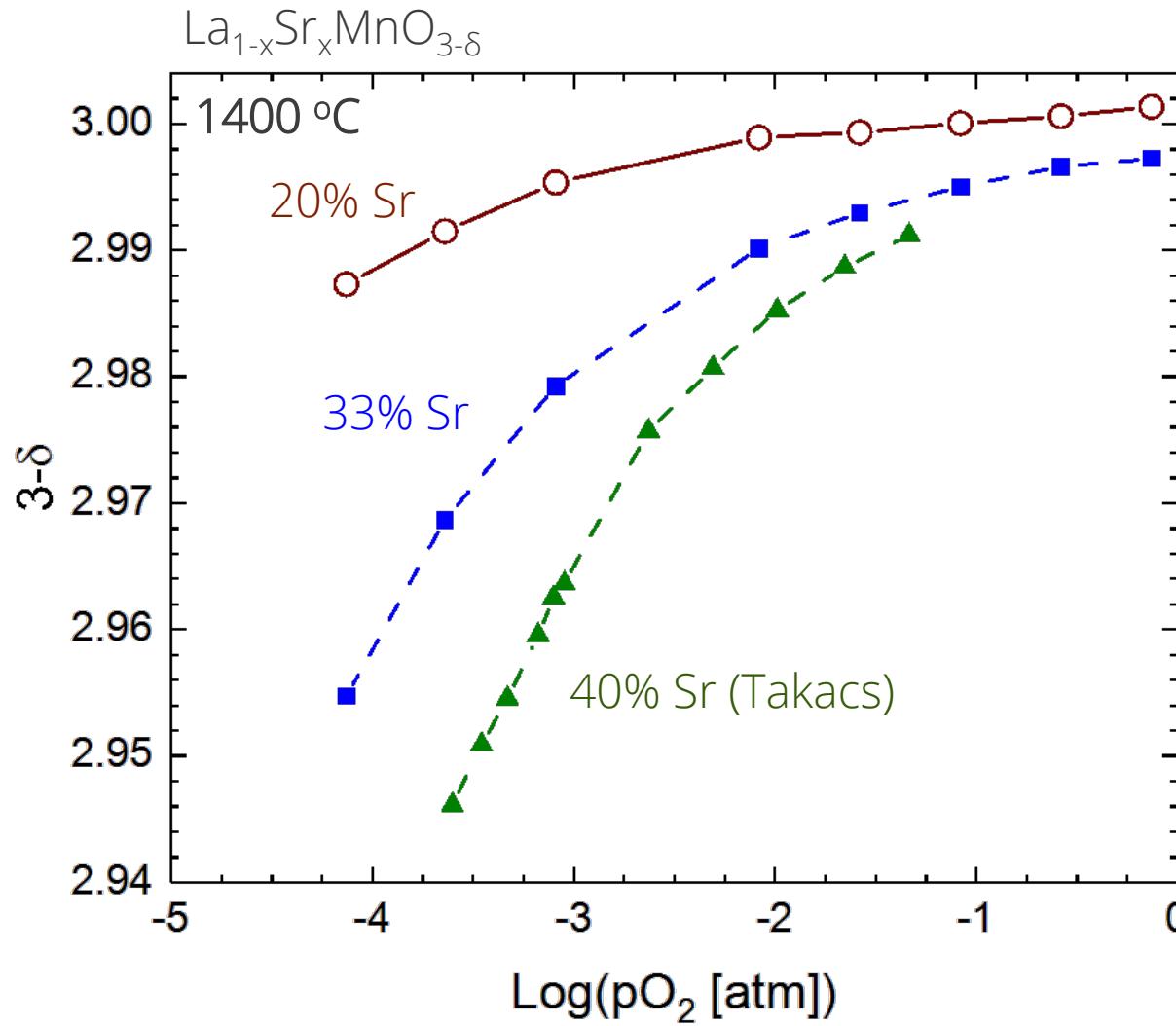
Increase in δ with temperature, as expected



*Mines: In collaboration with M. Sanders at CSM

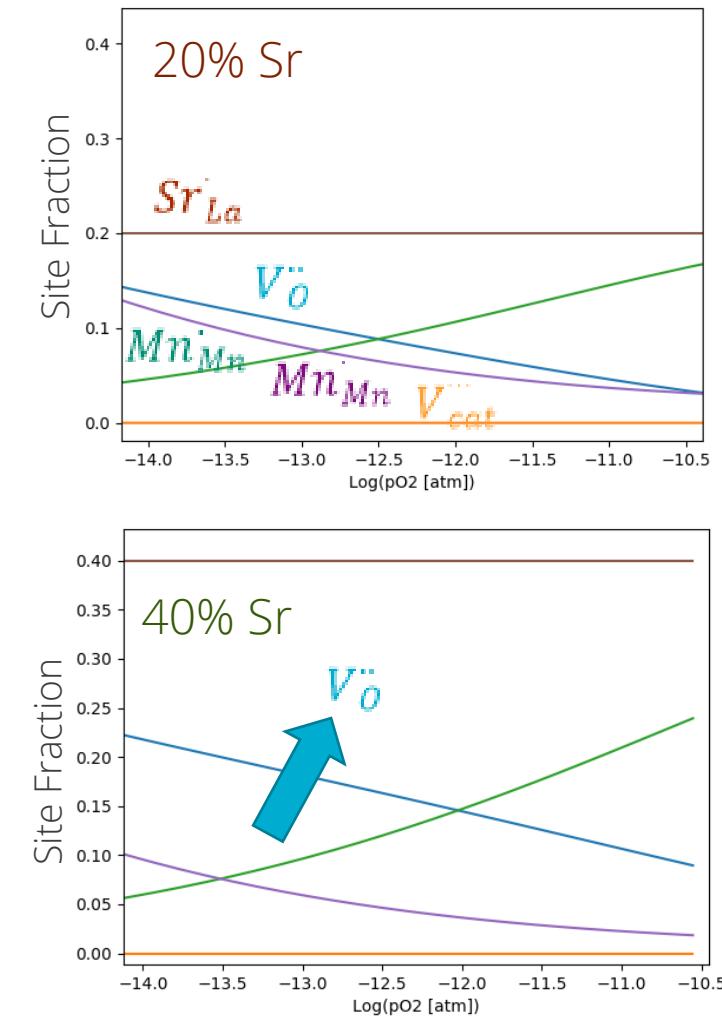
Future Work → Defect equilibria modeling

Impact of Acceptor Dopant Content on Reducibility



Takacs, et al., Acta Mater., 103 (2016) 700

*33% Sr in collaboration with X. Liu and J. Luo (WVU and UCSD)



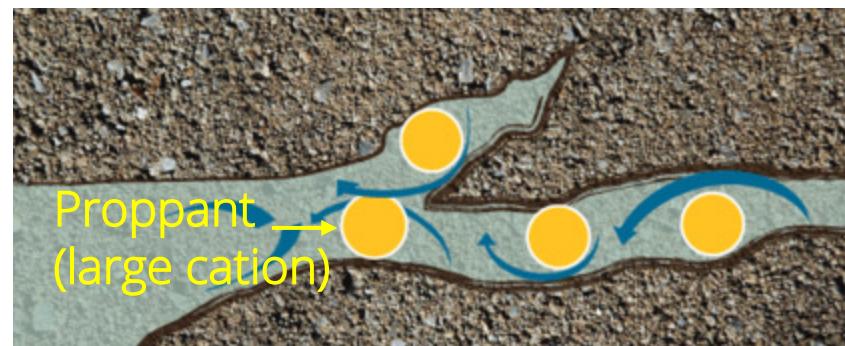
Increasing [Sr] → more oxygen vacancies due to charge compensation

Enhanced Reducibility of High Entropy Perovskite Oxide (HEPO)

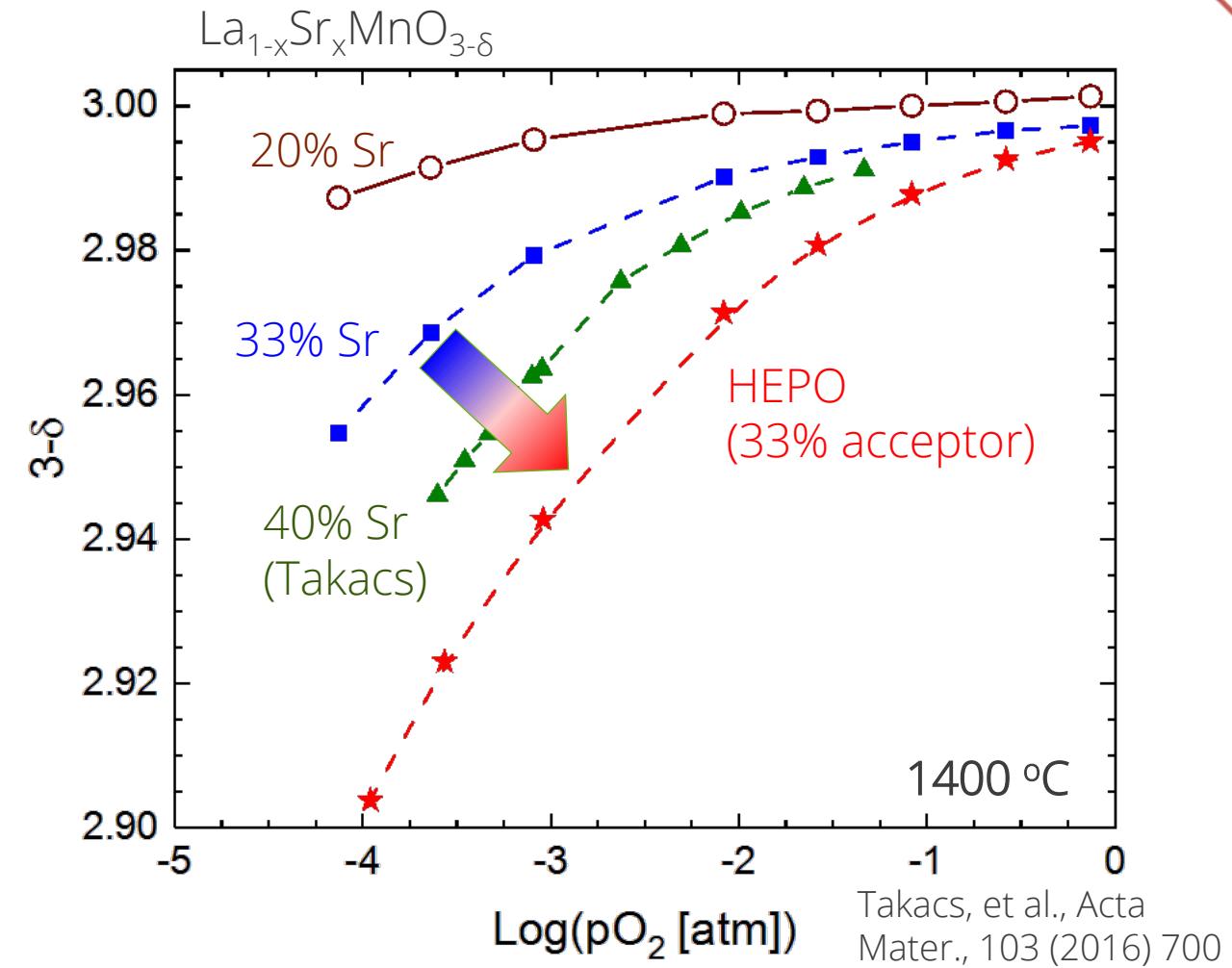


- HEPO and LSM33 have similar
 - Average A-site ionic radius
 - Lattice parameter
 - Total acceptor dopant content
- Even more reducible than LSM40

Explanation?: Large cations “fracked” LSM (small cation – oxygen bonds weakened)



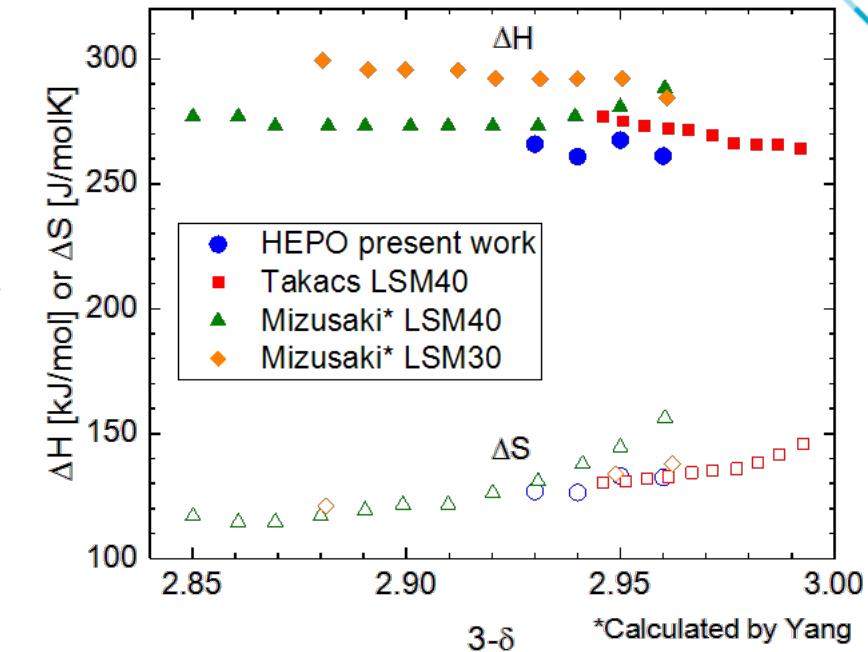
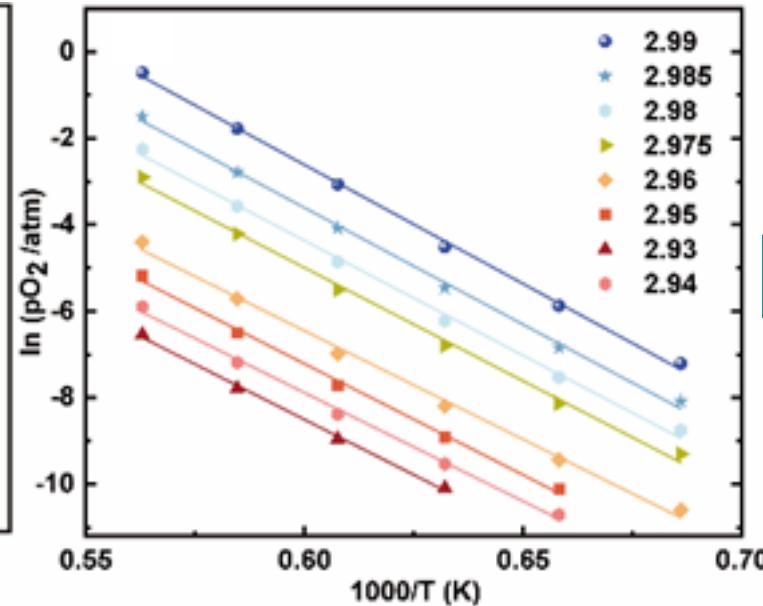
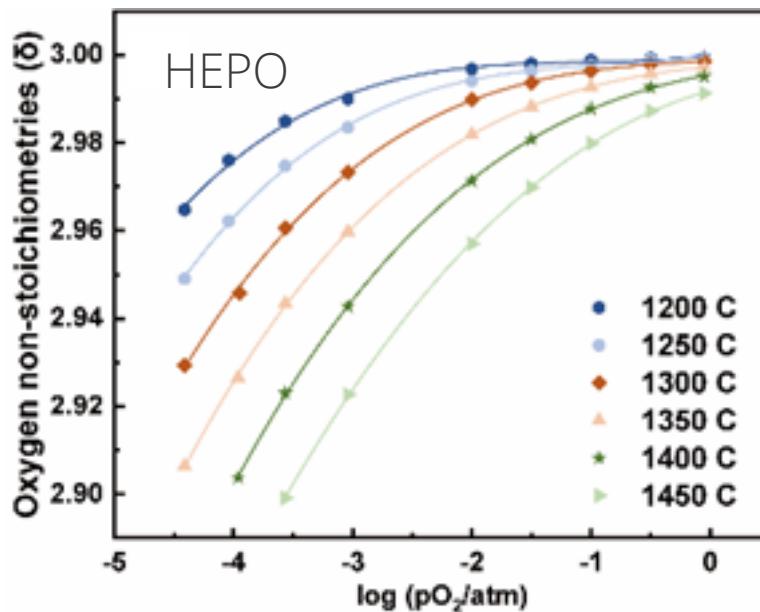
Shale Oil and Gas Handbook (2015)



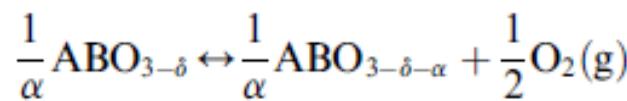
S. Bishop et al., ECS Trans. (2023)

*In collaboration with X. Liu (WVU) and J. Luo (UCSD)

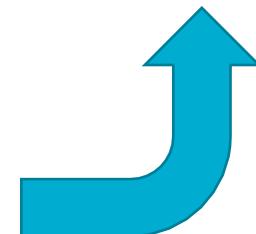
H_r Decrease for HEPO



Van't Hoff analysis



$$K_{\text{redox}} = (p\text{O}_2^*)^{1/2} = \exp\left(-\frac{\Delta H_{\text{redox}}^0 - T\Delta S_{\text{redox}}^0}{RT}\right)$$

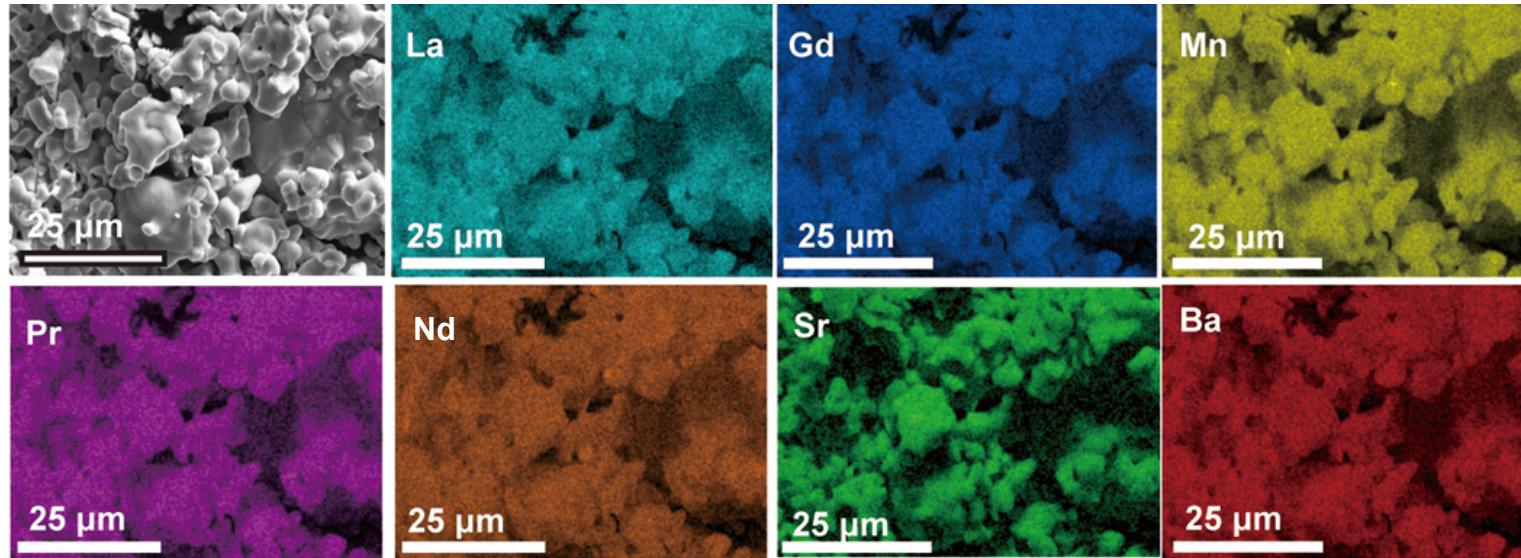


Takacs, et al., Acta Mater., 103 (2016) 700
 Mizusaki, et al., Solid State Ionics, 129 (2000) 163
 Yang, J. Mater. Chem. A, 2 (2014) 13612

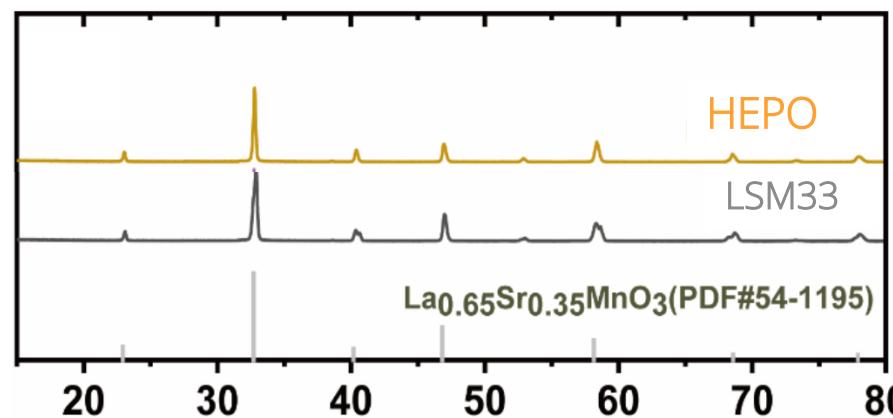
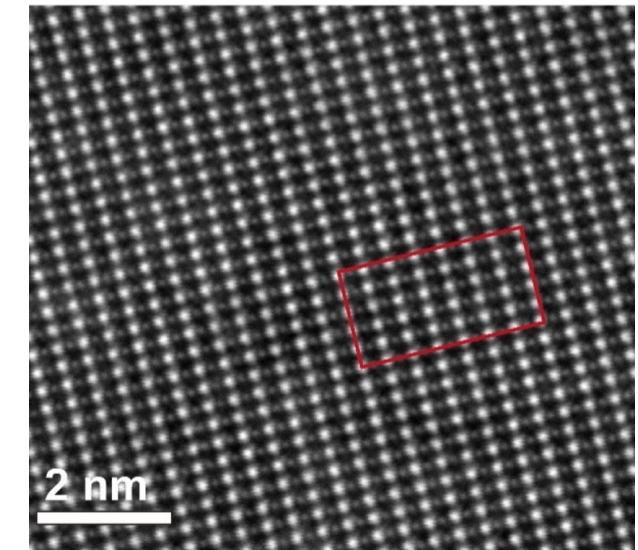
- *Lower H_r for HEPO compared to LSM*
- Entropy similar

Homogeneous Composition and Disordered Atomic Structure

SEM/EDS: No significant evidence of compositional heterogeneity

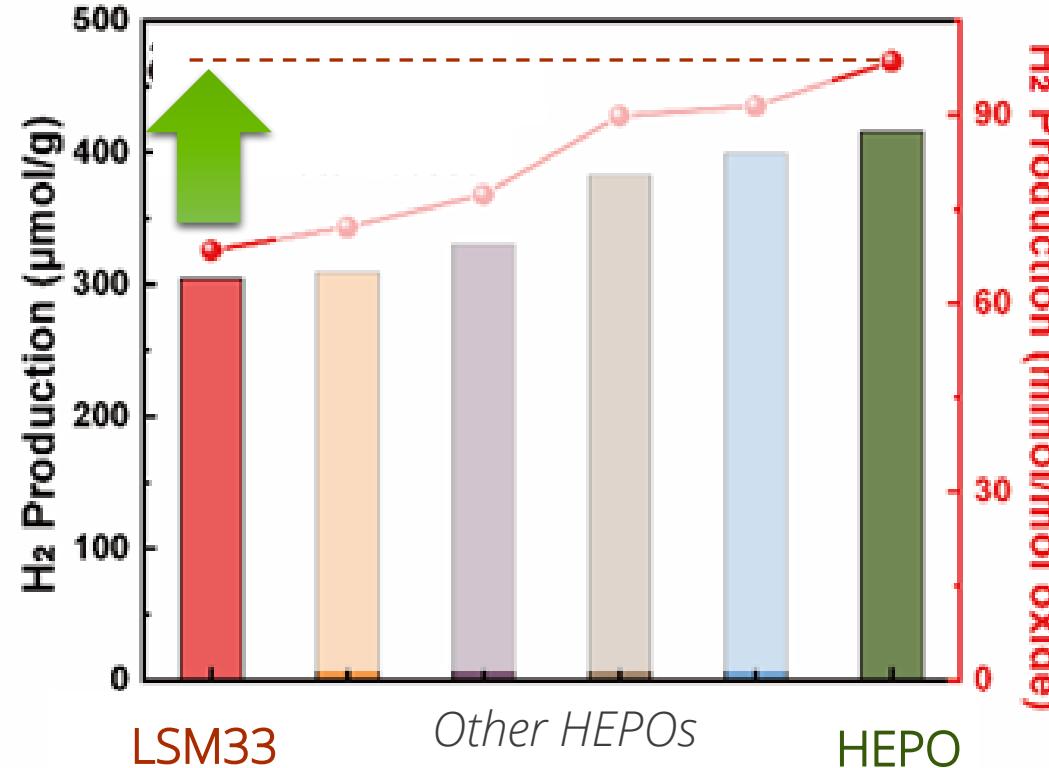


HAADF-STEM: No significant evidence of A-site cation ordering



- Expected perovskite structure of HEPO and LSM33
- HEPO ~0.2% smaller cell volume

Dramatic Increase in STCH H₂ Production



1350 °C N₂ reduction; 1100 °C 40% steam oxidation

46% increase in H₂ production with HEPO compared to LSM33!

Cijie Liu et al., J. Mater. Chem. A, 2024, 12, 3910-3922

Summary

- Presented exemplar STCH materials and key STCH metrics
 - Identified significant gap in H_2 production in low steam/ H_2 ratio materials
- Demonstrated successful water splitting with ML predicted compound
 - Improved thermal stability with Al addition
 - Competitive H_2 production to CeO_2 in low steam/ H_2
 - Despite severe morphological and phase changes, maintains H_2 production after many cycles
- Examined high temperature (>1200 °C) oxygen non-stoichiometry in LSM20
 - Experimental differences from thermodynamic predictions in high pO_2
- Discussed new $(La_{1/6}Pr_{1/6}Nd_{1/6}Gd_{1/6}Ba_{1/6}Sr_{1/6})MnO_{3-\delta}$ (HEPO) STCH material with much lower reduction enthalpy than LSM of same acceptor content



Acknowledgements

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- DOE, Office of Energy Efficiency and Renewable Energy (EERE), under the Agreement Number DE-EE0008839

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