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# PREDICTING AND EVALUATING NEW WATER SPLITTING MATERIALS FOR SOLAR THERMOCHEMICAL HYDROGEN PRODUCTION

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24<sup>th</sup> International Conference on Solid State Ionics, London, UK

3C2 Catalysis

11:15 am – 11:45 am, Wednesday, July 17, 2024 (Room Fleming) - Keynote

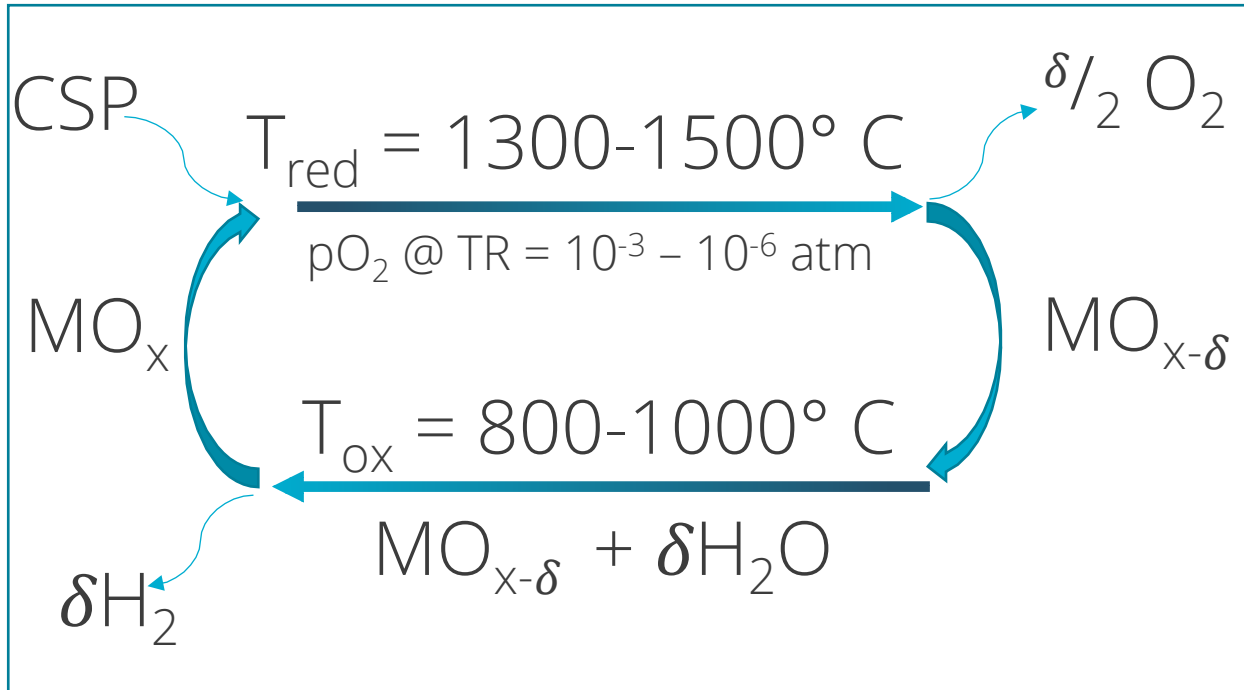
# Outline

- Solar thermo-chemical hydrogen production (STCH) background and the US DOE HydroGEN consortium
- Addressing needs for STCH community
  - Identification of benchmarking metrics and applying them to exemplar materials
  - New materials identification aided by computational discovery with machine learning approach
  - Revisiting oxygen non-stoichiometry in  $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_{3-\delta}$  at STCH reducing temperatures ( $>1200^\circ\text{C}$ ), and new High Entropy Perovskite Oxide (HEPO)

# Solar Thermo-Chemical Hydrogen Production (STCH)

*Sunlight (heat) + water  $\rightarrow$  Hydrogen*

National Solar Thermal Test Facility at Sandia National Labs

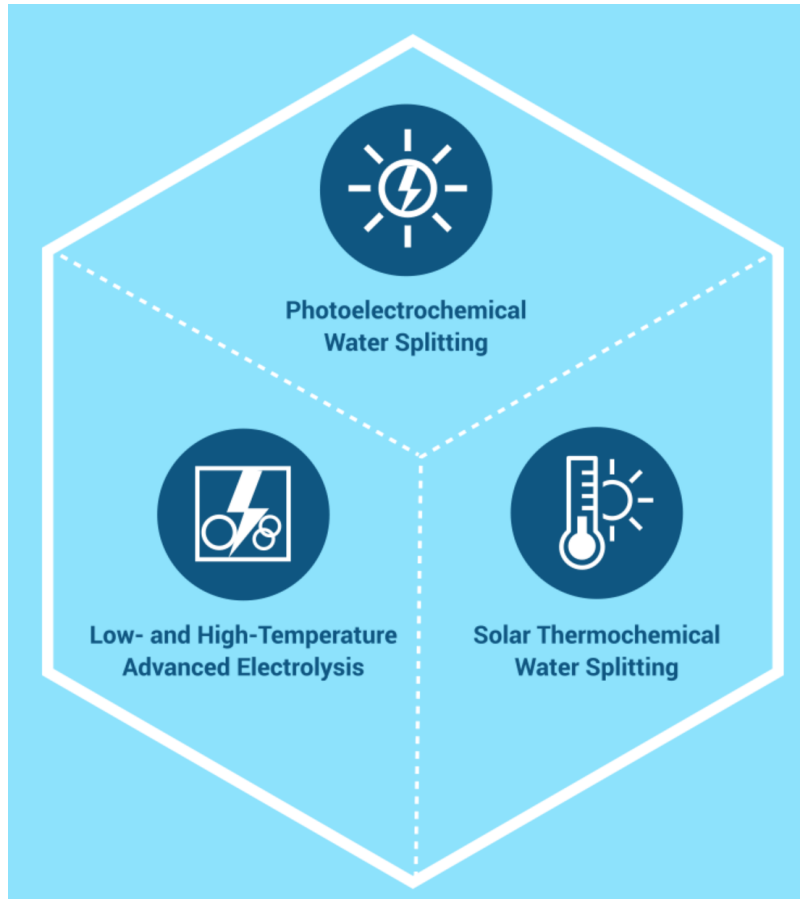


[energy.sandia.gov](http://energy.sandia.gov)

- Single phase materials to avoid cycling degradation
- Oxides that “breathe” oxygen needed  $\rightarrow$  non-stoichiometric oxides
- Goldilocks reduction enthalpy ( $H_r$ )
- Also use heat from other sources  $\rightarrow$  *Thermo-chemical hydrogen production (TCH)*

# Department of Energy HydroGEN Consortium (H2AWSM.org)

*Research focus areas*



Leverage capabilities at national laboratories

- National Renewable Energy Laboratory (Lead)
- Lawrence Berkeley National Laboratory
- Sandia National Laboratories
- Idaho National Laboratory
- Lawrence Livermore National Laboratory

Collaboration with industry and academia seedlings supported by HydroGEN funding opportunities

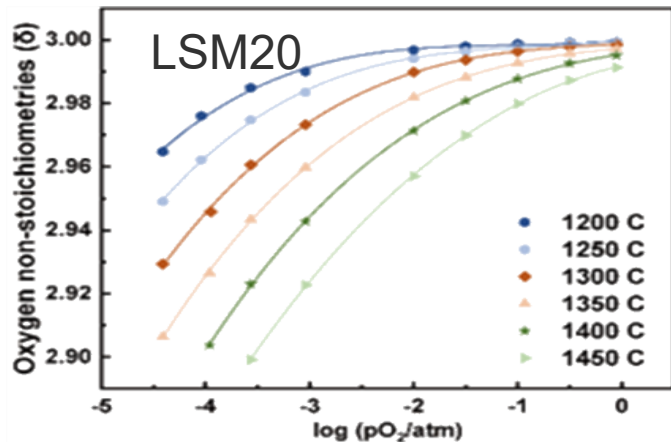
Sandia node capabilities used for this presentation:

- Thermo-gravimetric analysis (Sean Bishop)
- STCH reactor (Tony McDaniel)
- Electron microscopy and composition analysis (Josh Sugar)

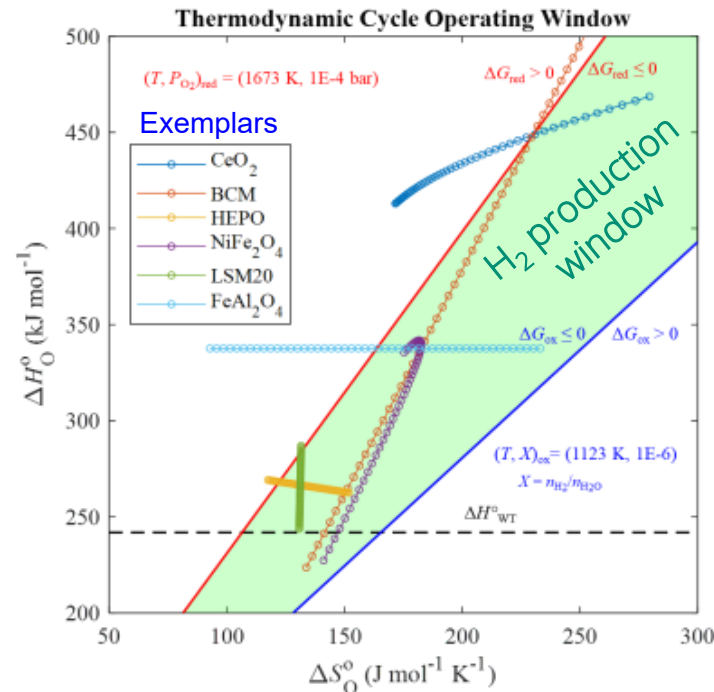
# Exemplar Material Viability Study

- Define STCH metrics
- Benchmark exemplar materials against state of the art ( $\text{CeO}_2$ )
- Identify technology gaps

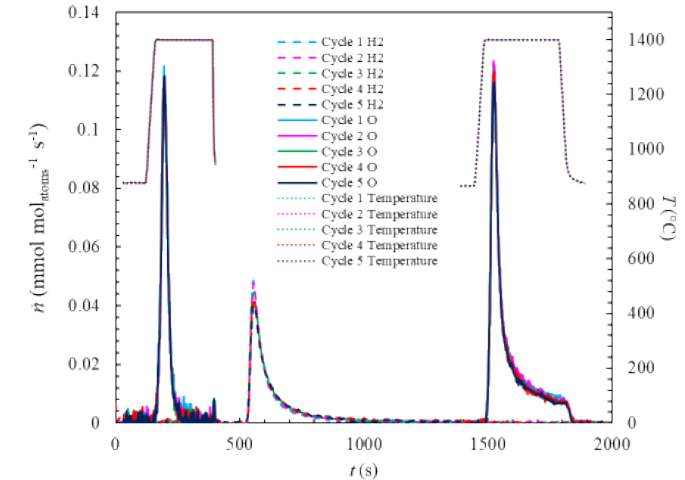
## Thermodynamic parameters (from thermo-gravimetric analysis)



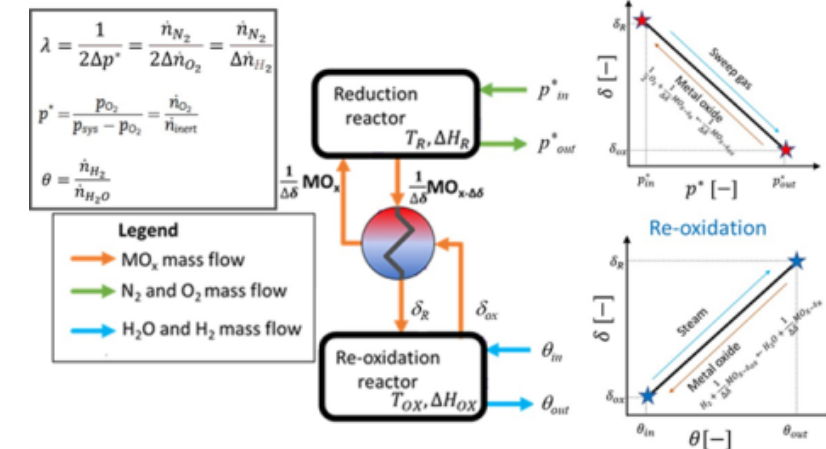
BCM:  $\text{BaCe}_{0.25}\text{Mn}_{0.75}\text{O}_3$   
 HEPO:  $\text{La}_{1/6}\text{Pr}_{1/6}\text{Nd}_{1/6}\text{Gd}_{1/6}\text{Ba}_{1/6}\text{Sr}_{1/6}\text{MnO}_3$   
 LSM20:  $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_3$



## Hydrogen production and kinetic parameter (from flow reactor)



## Cycle efficiency estimation



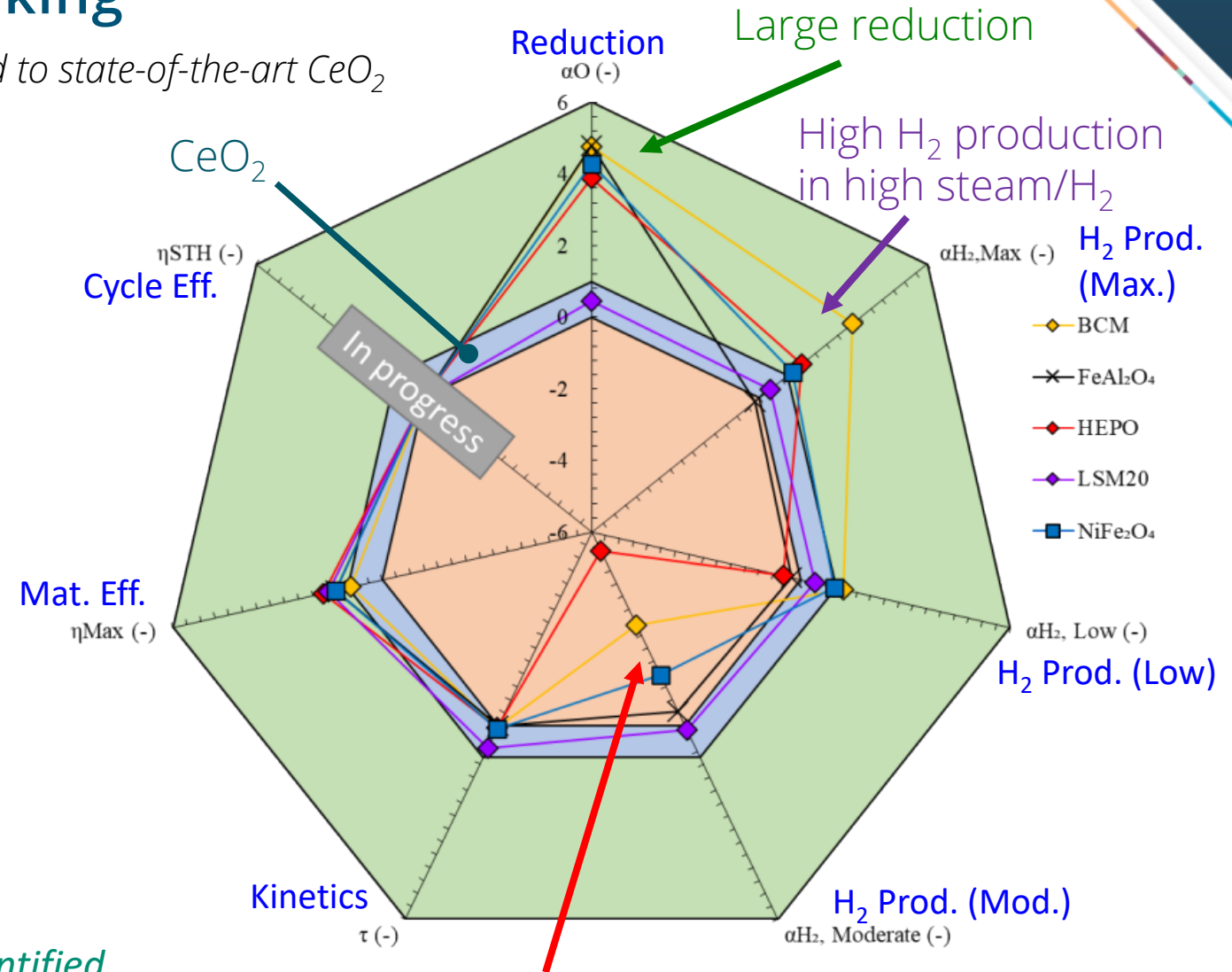


# Metrics and Exemplar Benchmarking

Exemplars normalized to state-of-the-art  $\text{CeO}_2$

## Metrics

	Descriptor	Target Values
Cycle Efficiency (STH)	Solar-to-hydrogen conversion efficiency derived from detailed cycle analysis using a thermodynamic model based on specific plant operational assumptions	$\eta_{\text{STH}} > 26\%$
Material Efficiency	$\frac{\Delta G_{\text{WS}}^o}{\Delta H_2^o}$ is the maximum possible thermal efficiency of the two-step process. ( $\Delta G_{\text{WS}}^o$ evaluated at 25 °C)	$\eta_{\text{Max}} > 50\%$
Reduction Capacity	mmol O / mol atom in solid reduced @ neutral low condition	$\alpha_{\text{O}} > 5$
STCH Capacity (Maximum Yield)	mmol $\text{H}_2$ / mol atom in solid reduced @ neutral low condition, oxidized in pure $\text{H}_2\text{O}$ @ optimal $T_{\text{OX}}$ for material	$\alpha_{\text{H}_2, \text{Max}} > 5$
STCH Capacity (Low Yield)	mmol $\text{H}_2$ / mol atom in solid reduced @ neutral low condition, oxidized in steam-to-fuel ratio $\text{H}_2\text{O}/\text{H}_2 = 1000$ @ optimal $T_{\text{OX}}$ for material	$\alpha_{\text{H}_2, \text{Low}} > 2.5$
STCH Capacity (Moderate Yield)	mmol $\text{H}_2$ / mol atom in solid reduced @ neutral low condition, oxidized in steam-to-fuel ratio $\text{H}_2\text{O}/\text{H}_2 = 100$ @ optimal $T_{\text{OX}}$ for material	$\alpha_{\text{H}_2, \text{Mod}} > 1$
Kinetic Performance	Time to 90% of $\alpha_{\text{H}_2, \text{Max}}$ in pure $\text{H}_2\text{O}$ at optimal $T_{\text{OX}}$ for specific material in a dispersed powder configuration	$\tau > 0.20$

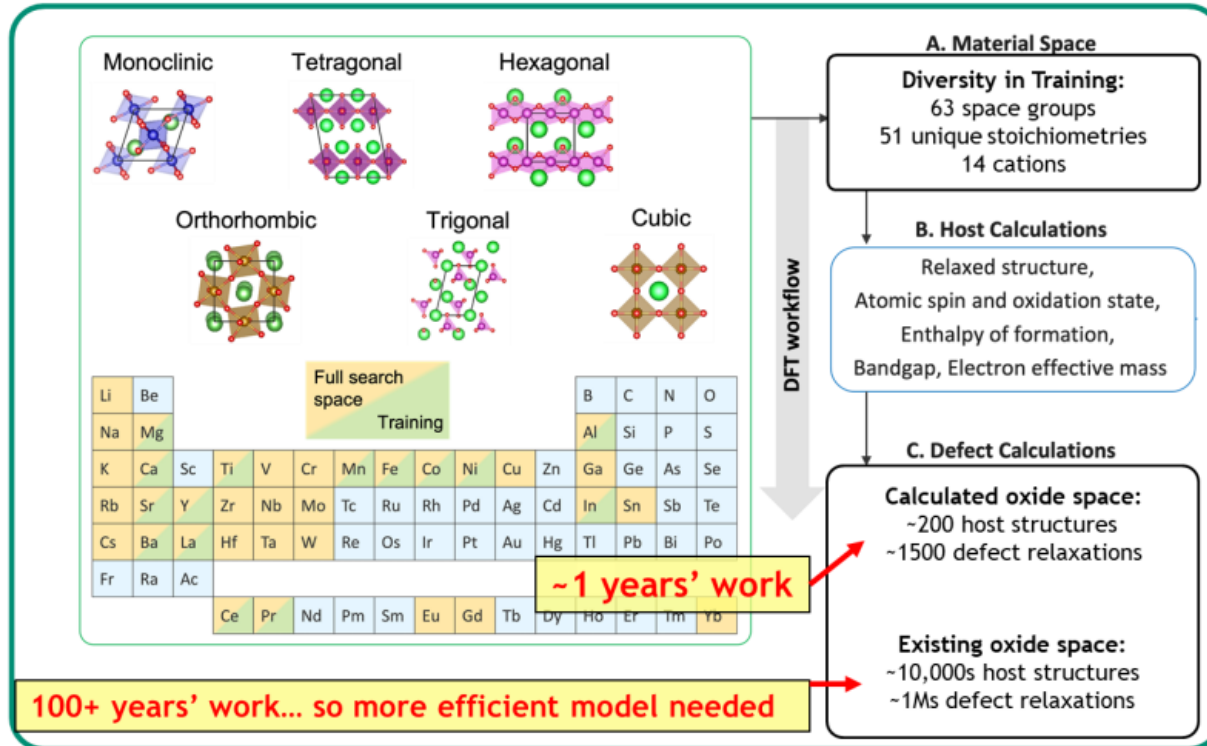


- Evaluation framework created and metrics identified
- Weakness of exemplars in low steam/ $\text{H}_2$  ratio → critical need for new materials

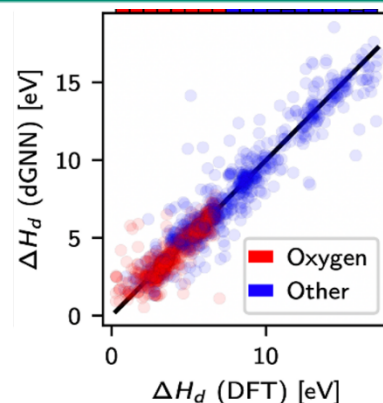
...but  $\text{H}_2$  "consumption" in low steam/ $\text{H}_2$  ratio → only LSM20 competitive with  $\text{CeO}_2$

# Discovery of New STCH Materials Aided by Machine Learning

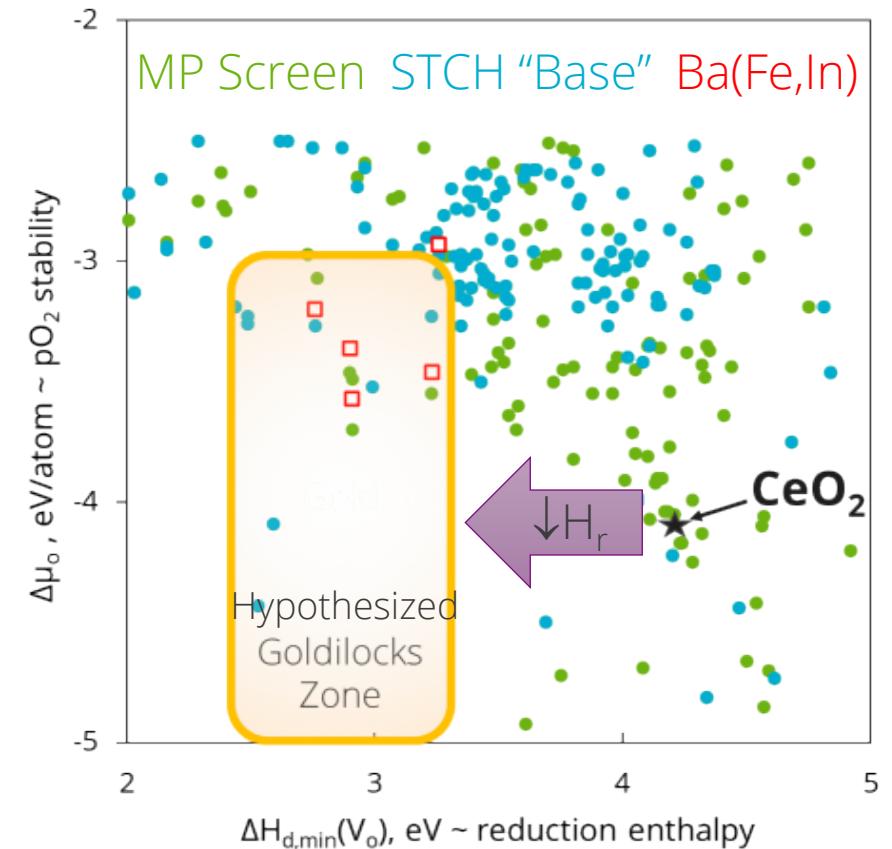
First-principles DFT workflow is robust but costly (using NRELMatDb hosts)



ML screens 10,000's of MP structures in minutes that would take 1,000's of DFT months



Expected  $\Delta H_{0,d}$  MAE for unseen compounds < 450 meV (threshold for ML to be predictive).

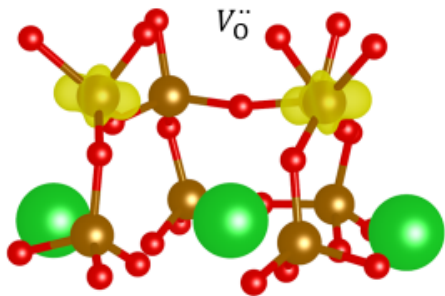


→ Examples: Identified  $Ba(Fe,In)_2O_4$  and  $(Ba,Sr)_6(La,Y)_2Fe_4O_{15}$

# BaFe<sub>2</sub>O<sub>4</sub>: V<sub>O</sub><sup>••</sup> Point Defect Mechanism, and Al Addition to Increase Temperature Stability



Orthorhombic (Bb21m) phase



DFT → oxygen vacancy preferred vs. cation defects

Melting of BFO in low pO<sub>2</sub> mitigated by Al substitution!

1400 °C in Ar (~20 ppm O<sub>2</sub>)



BaAl<sub>0.4</sub>Fe<sub>1.6</sub>O<sub>4</sub> – Hexagonal (P6<sub>3</sub>) phase

→ Enables higher reduction temperature and resistance to densification

Higher melting point expected with Al

- BaAl<sub>2</sub>O<sub>4</sub> (hexagonal P6<sub>3</sub>, stuffed tridymite) melting point ~1820 °C
- MgAl<sub>2</sub>O<sub>4</sub> (cubic Fd-3m, spinel) melting point ~2130 °C

BaAl<sub>0.4</sub>Fe<sub>1.6</sub>O<sub>4</sub> stable at 1450 °C in air!

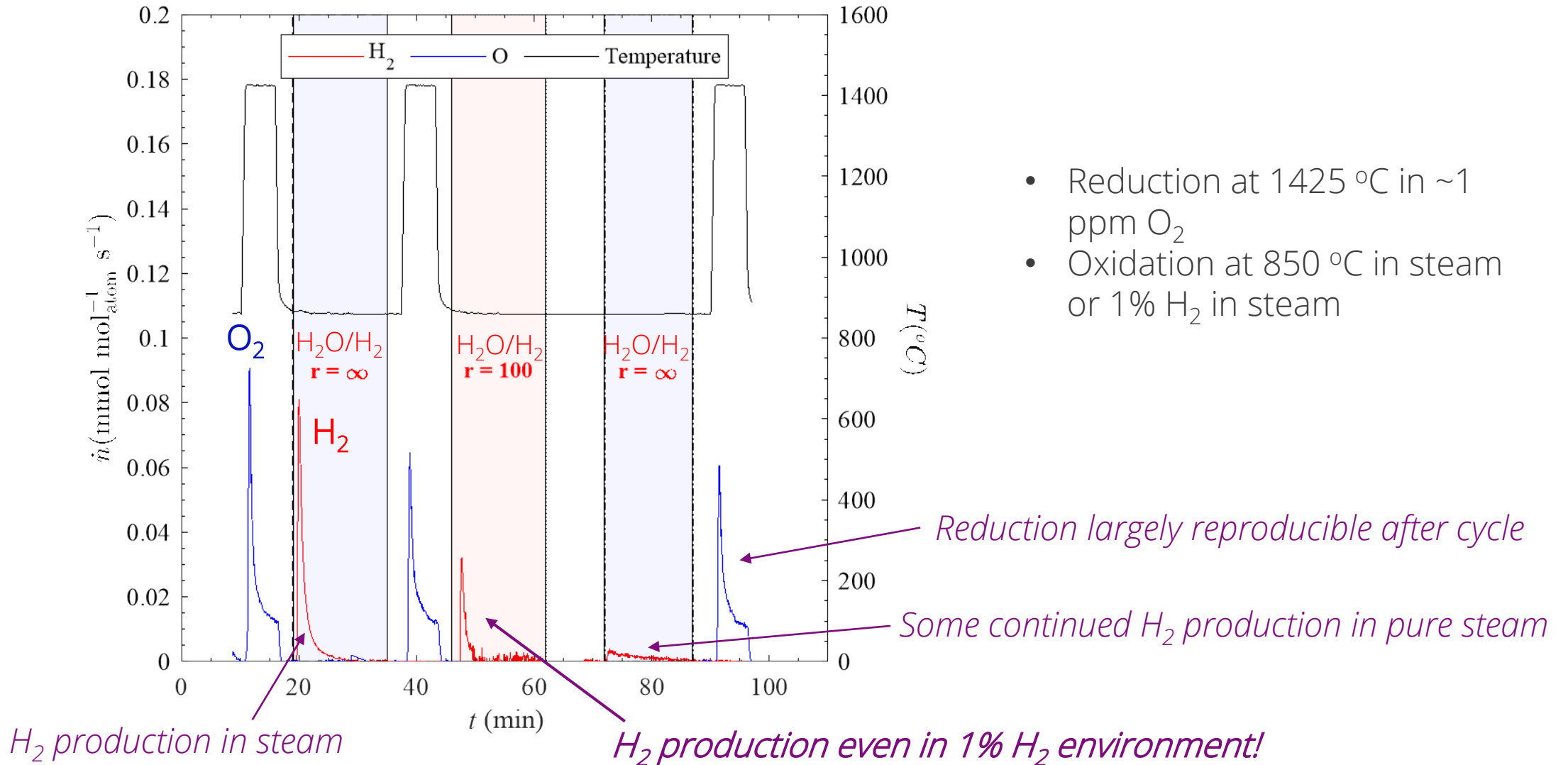
BaFe<sub>2</sub>O<sub>4</sub> T<sub>melt,air</sub> ~1420 °C



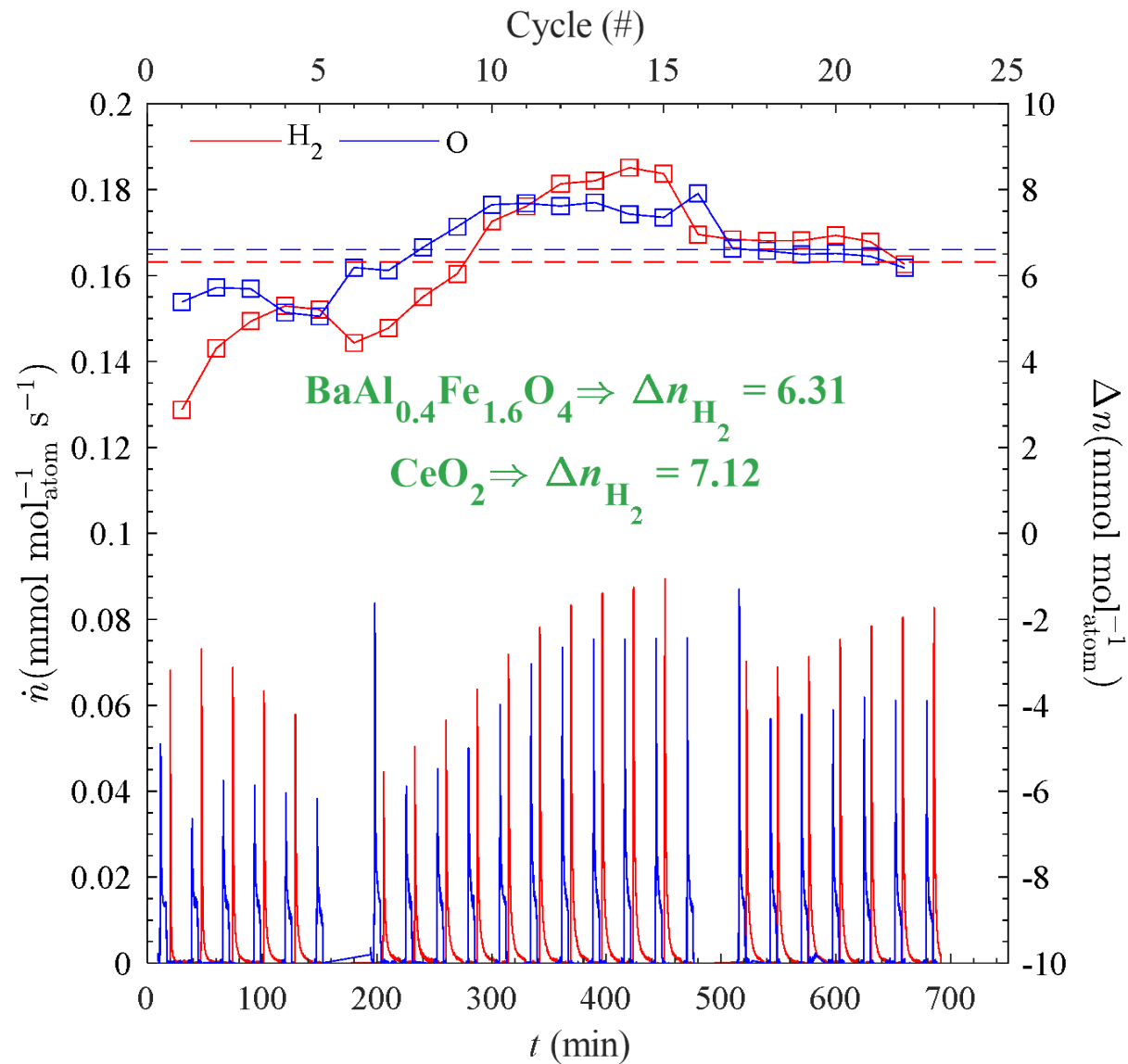


# BaAl<sub>0.4</sub>Fe<sub>1.6</sub>O<sub>4</sub>: Water Splitter in Low Steam/H<sub>2</sub>!

H<sub>2</sub> production measurement using flow reactor



# H<sub>2</sub> Production Over 20 Cycles



- Reduction at 1425 °C in ~1 ppm O<sub>2</sub>
- Oxidation at 850 °C in steam

*Comparable H<sub>2</sub> production performance to CeO<sub>2</sub> retained for 20 cycles!*

# Severe Change in Morphology and Phase After Cycling

Initial powder



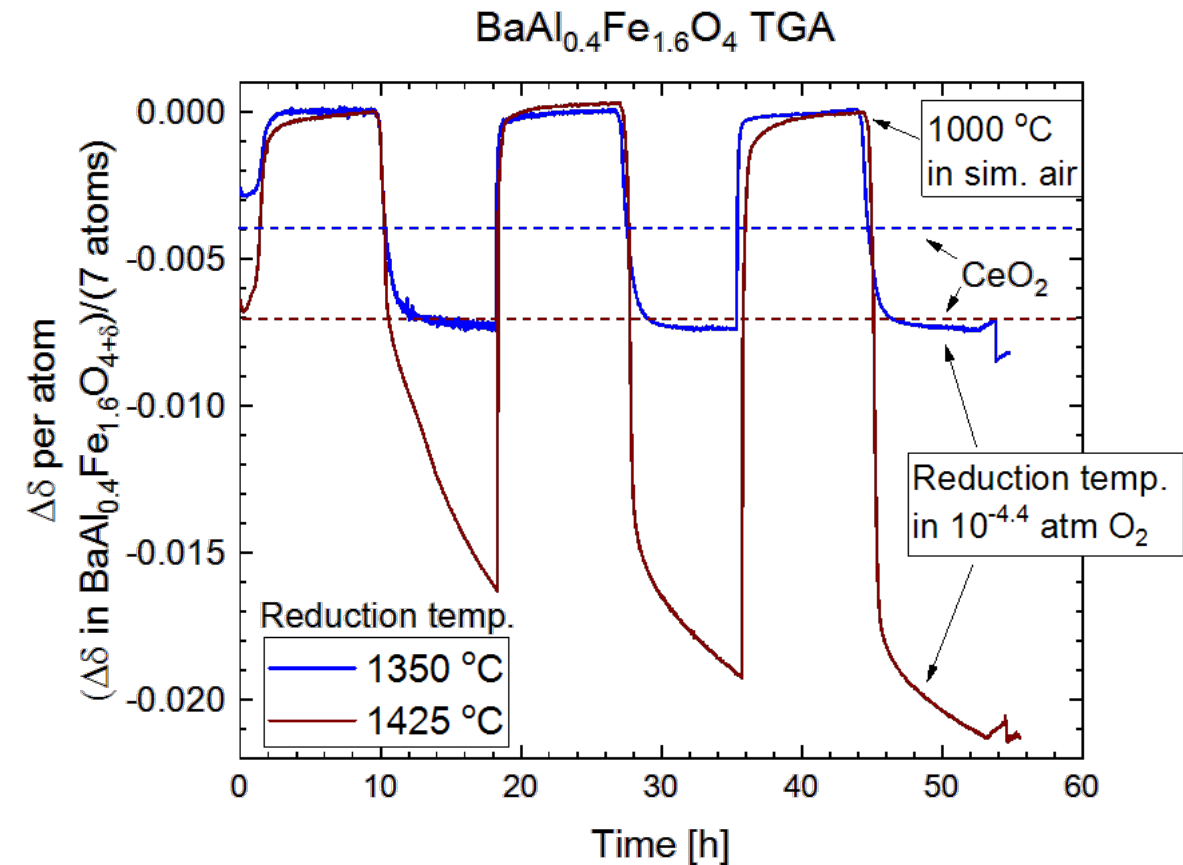
After 20 redox cycles



*Post-cycled sample initial diffraction analysis at SLAC*

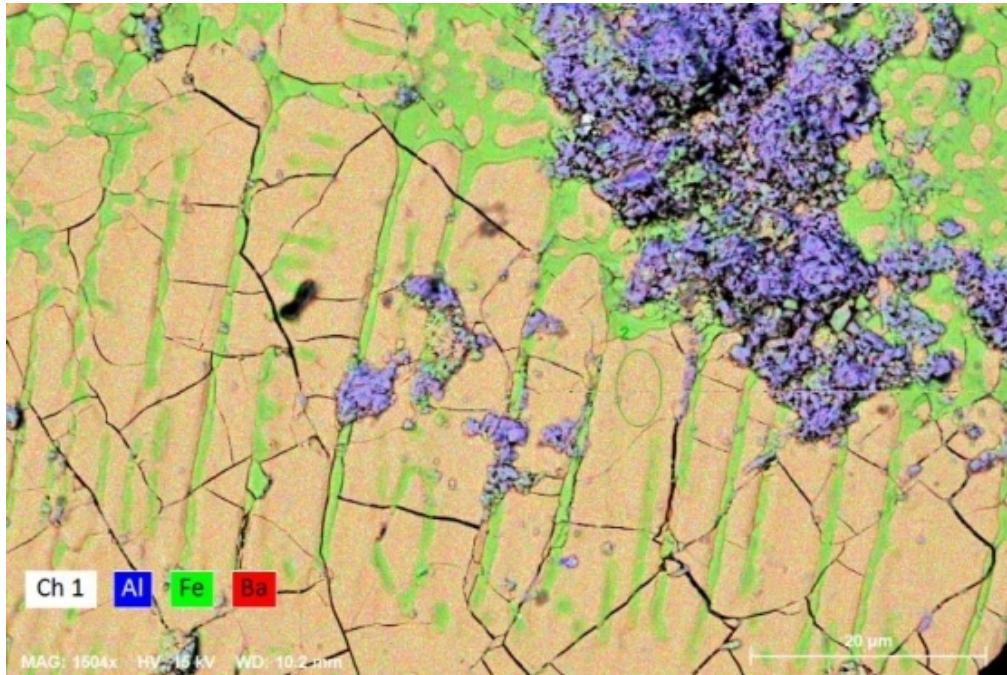
- $\text{BaAl}_2\text{O}_4$  (P6<sub>3</sub>) is 20.34 wt% → starting phase
  - $\text{BaFe}_2\text{O}_4$  (Cmc21) is 75.17 wt%
  - $\text{BaFeO}_3$  (P6<sub>3</sub>/mmc) is 4.48 wt%
- } Decomposition products

*Long-term reduction at 1425 °C  
→ continuous oxygen release*



# Detailed Compositional Analysis – 3-phases

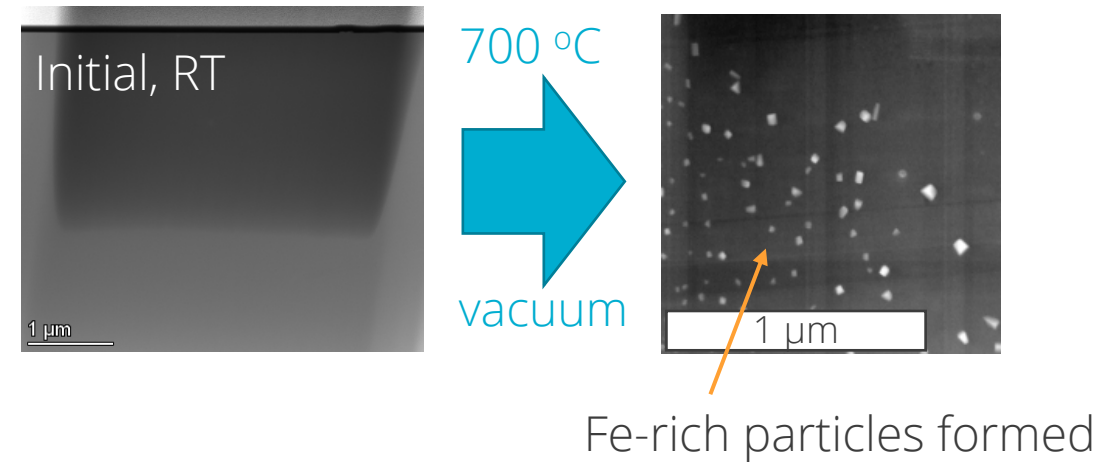
EDS map of post-20 cycled sample



- Purple phase consistent with  $\sim\text{BaAlFeO}_4$  (the expected  $\text{P6}_3$  structure)
- Tan phase consistent with  $\sim\text{BaFeO}_3$ , with possible melting
- Green phase consistent with  $\sim\text{BaFe}_2\text{O}_4$ , with possible melting

*In situ* reduction in TEM  $\rightarrow$  evolution of decomposition

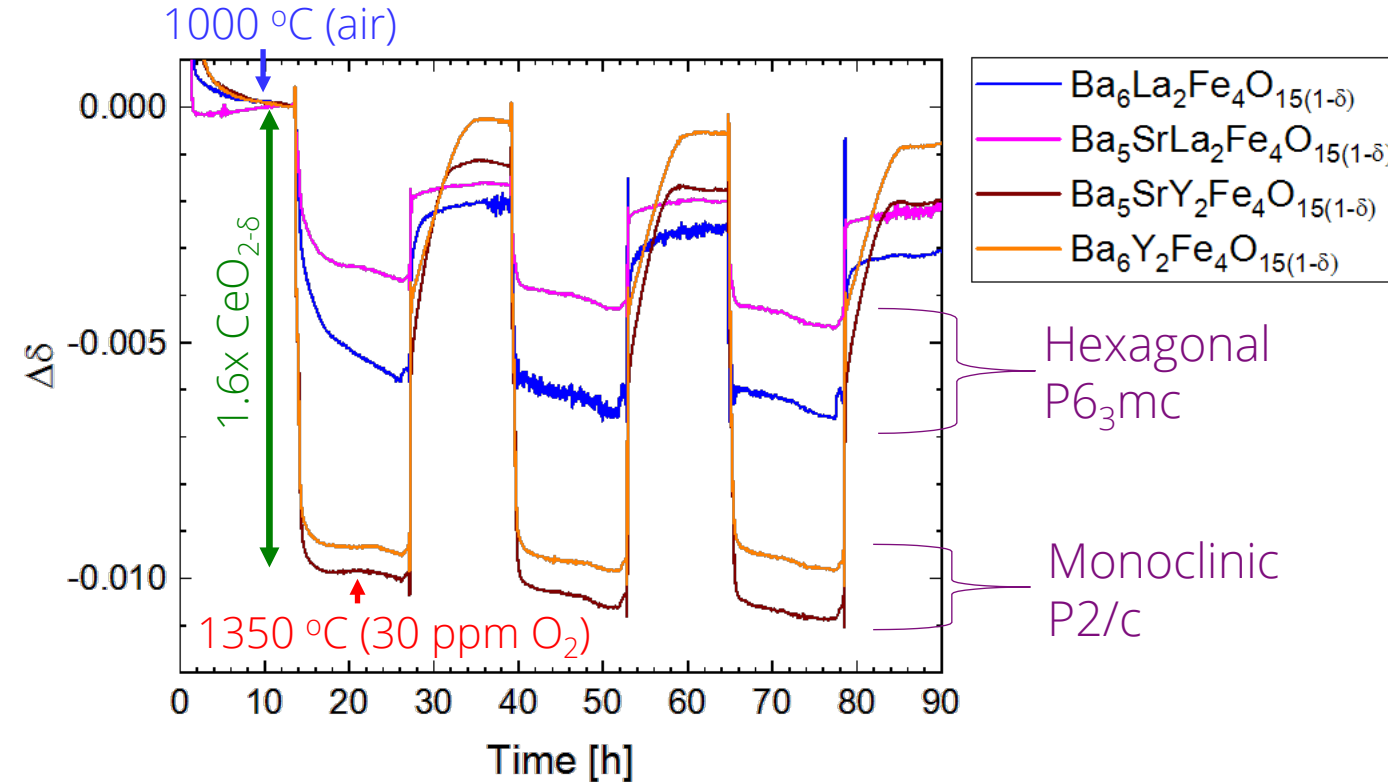
FIB cross-section of  $\text{BaAl}_{0.4}\text{Fe}_{1.6}\text{O}_4$



*Future Work:*  
 $\rightarrow$  investigating role of each phase in water splitting and durability  
 $\rightarrow$  HTXRD in simulated water splitting conditions

# $(\text{Ba,Sr})_6\text{La}_2\text{Fe}_4\text{O}_{15}$ and $(\text{Ba,Sr})_6\text{Y}_2\text{Fe}_4\text{O}_{15}$ ML Predicted Water Splitters

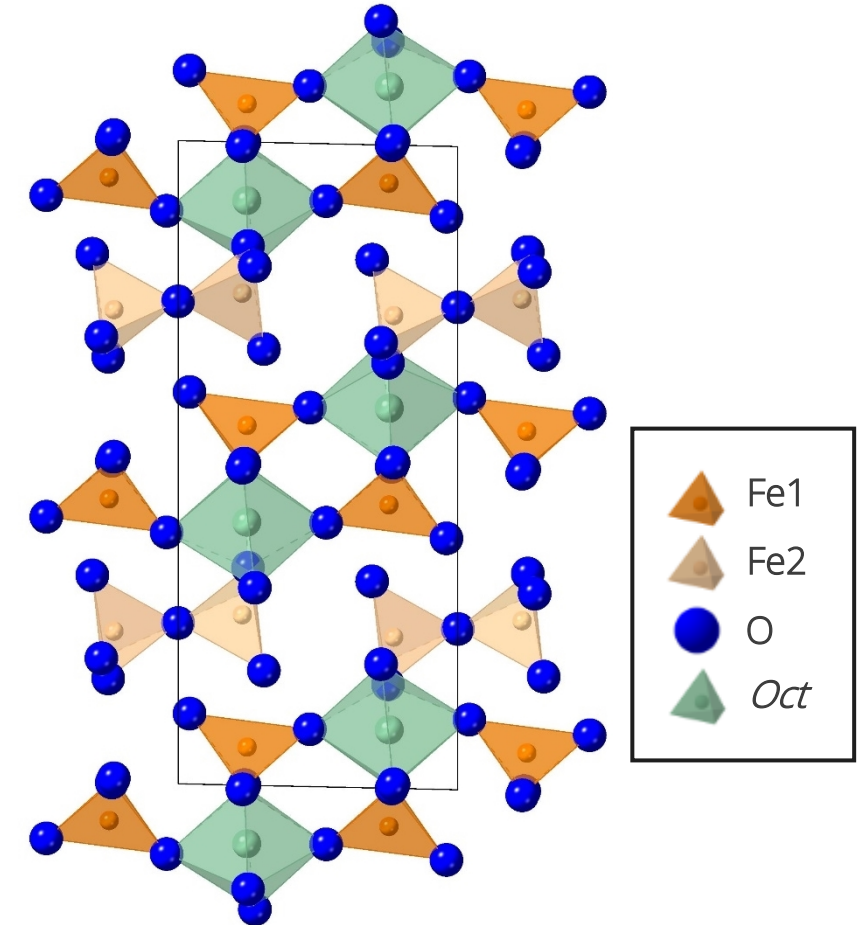
Generalized monoclinic structure (excluding Ba,Sr)  
Fe apex sharing dimers (not in hexagonal structure)



STCH screening protocol<sup>1</sup> →  $\text{Ba}_6\text{Y}_2\text{Fe}_4\text{O}_{15}$  is best

<sup>1</sup>Sanders et al., Front. Energy Res. 10:856943 (2022)

Monoclinic →  $\Delta\delta > \text{CeO}_2$

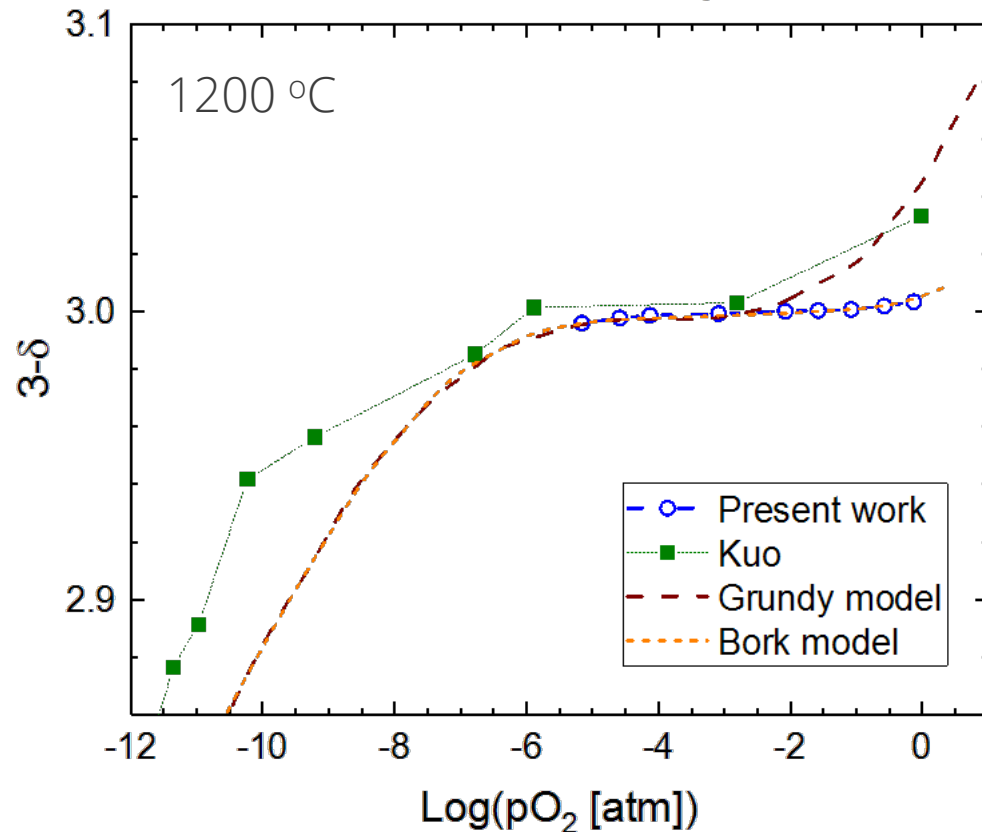


Future Work → Flow reactor measurements



# Re-Visiting $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_3$ : Oxygen Stoichiometry at High Temp.

Limited  $\delta$  data above 1200 °C  $\rightarrow$  evaluate at higher temperatures (e.g., STCH conditions)



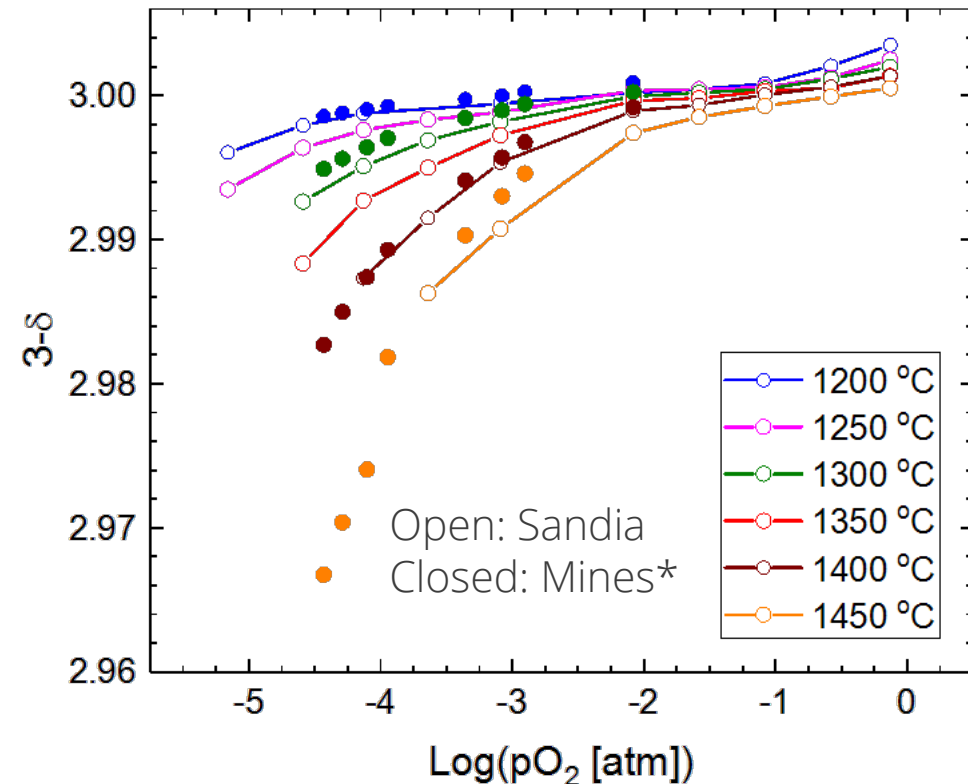
Less super-stoichiometry than previously estimated, in agreement with Bork model

Kuo et al., J. Sol. State Chem., 83 (1989) 52

Bork et al., Adv. Energy Mater. 7 (2017) 1601086

Grundy et al., CALPHAD 28 (2004) 191

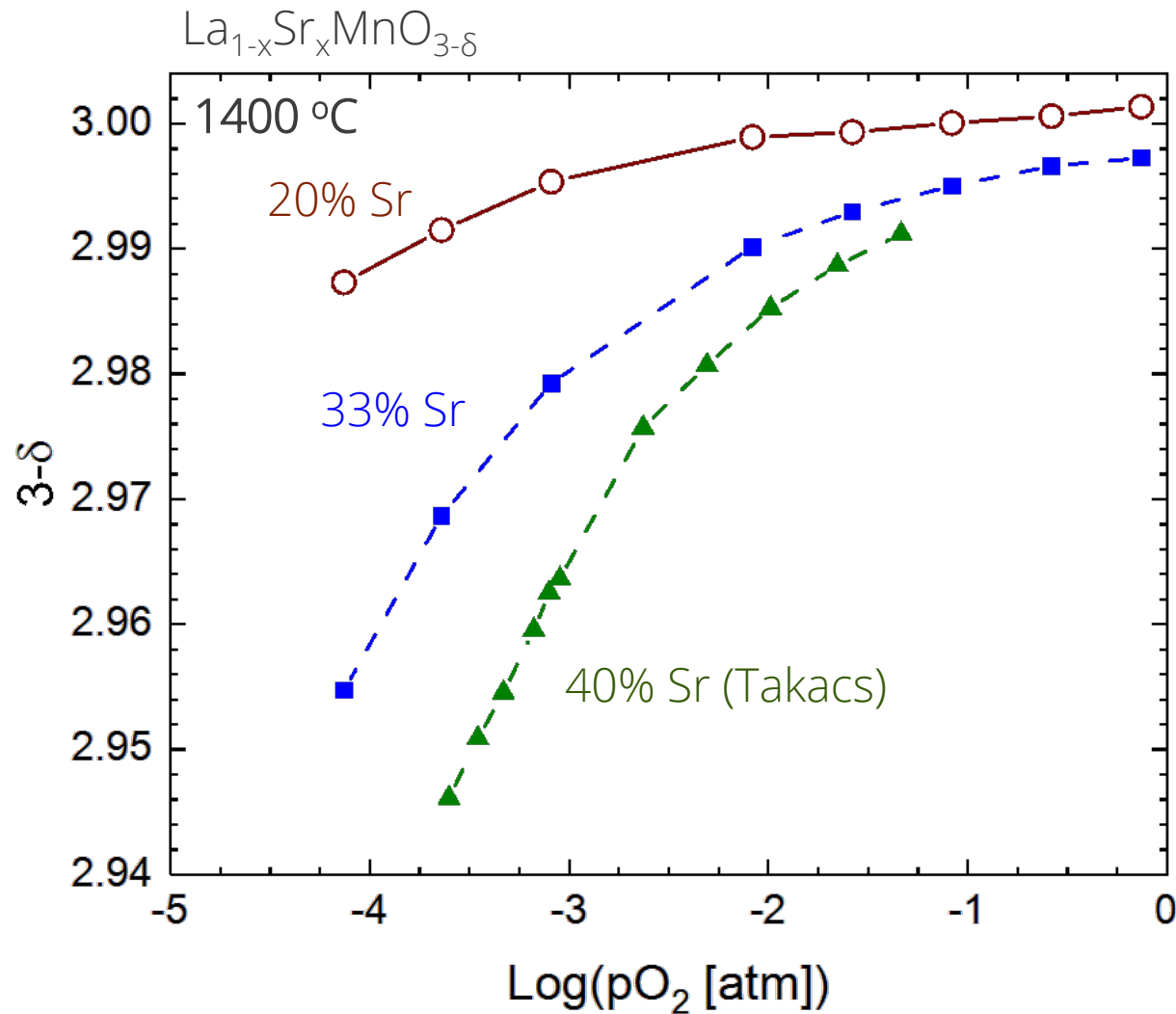
Increase in  $\delta$  with temperature, as expected



\*Mines: In collaboration with M. Sanders at CSM

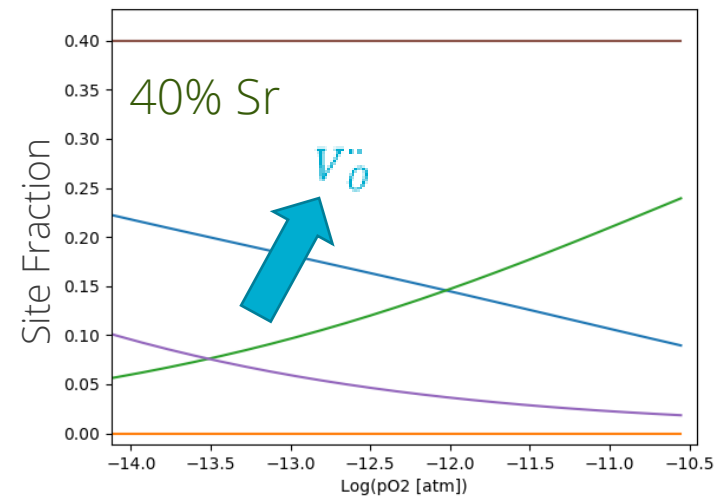
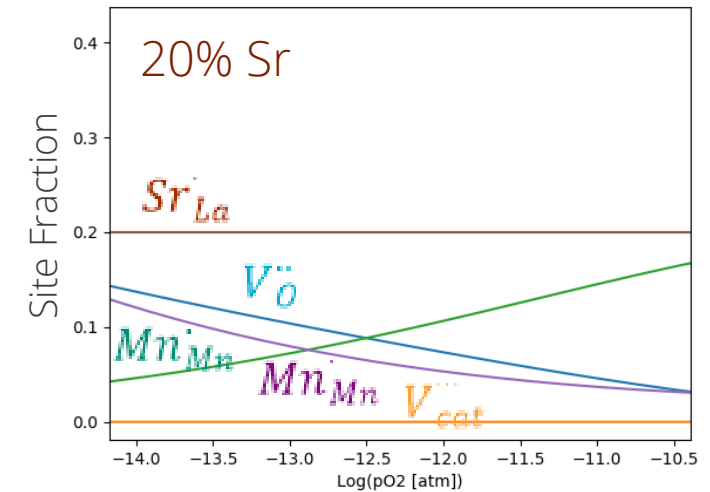
*Future Work  $\rightarrow$  Defect equilibria modeling*

# Impact of Acceptor Dopant Content on Reducibility



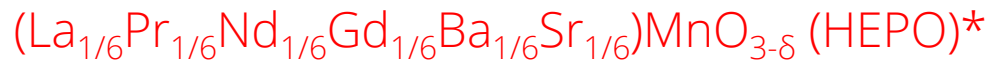
Takacs, et al., Acta Mater., 103 (2016) 700

\*33% Sr in collaboration with X. Liu and J. Luo (WVU and UCSD)



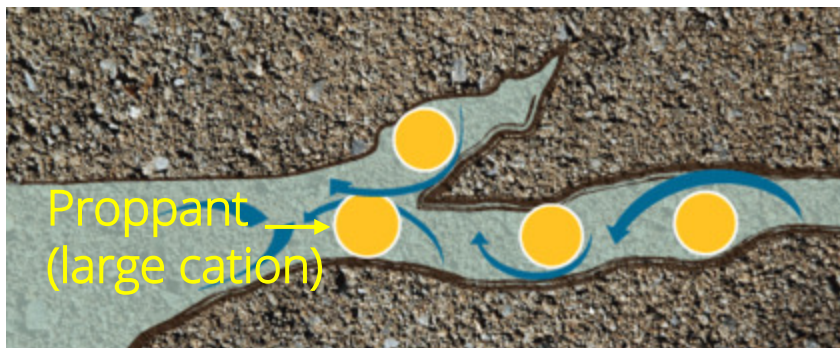
Increasing [Sr] → more oxygen vacancies due to charge compensation

# Enhanced Reducibility of High Entropy Perovskite Oxide (HEPO)

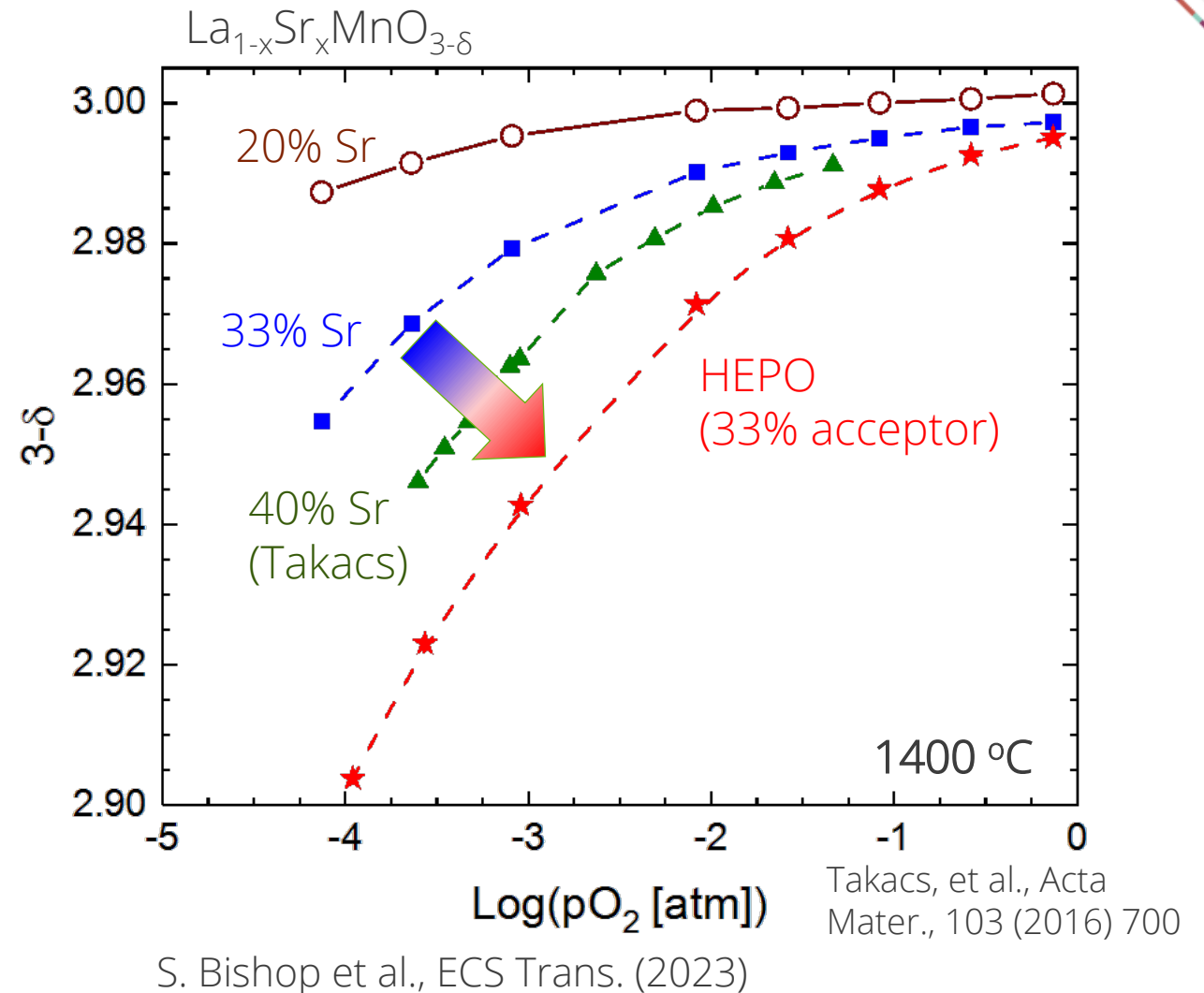


- HEPO and LSM33 have similar
  - Average A-site ionic radius
  - Lattice parameter
  - Total acceptor dopant content
- Even more reducible than LSM40

*Explanation?: Large cations “fracked” LSM (small cation – oxygen bonds weakened)*

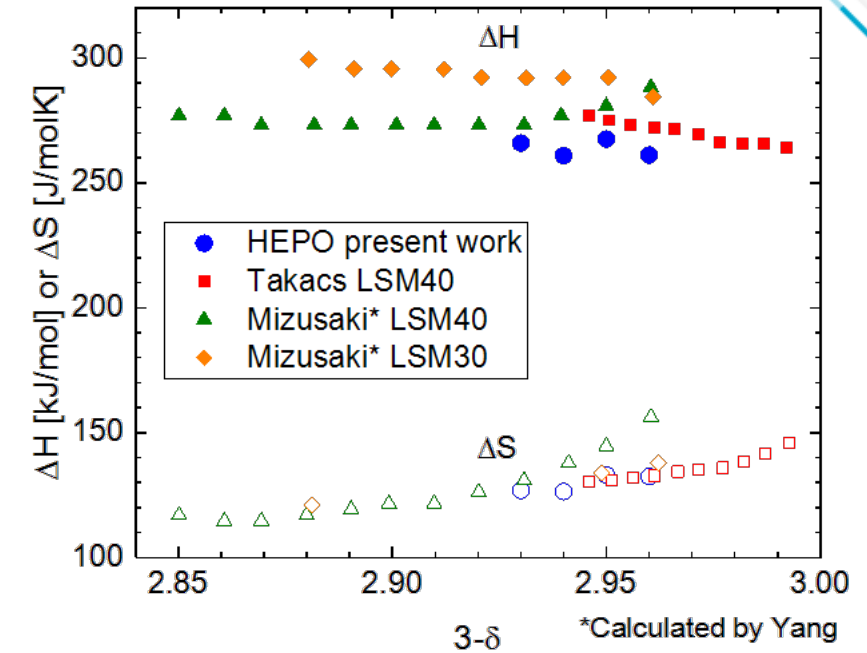
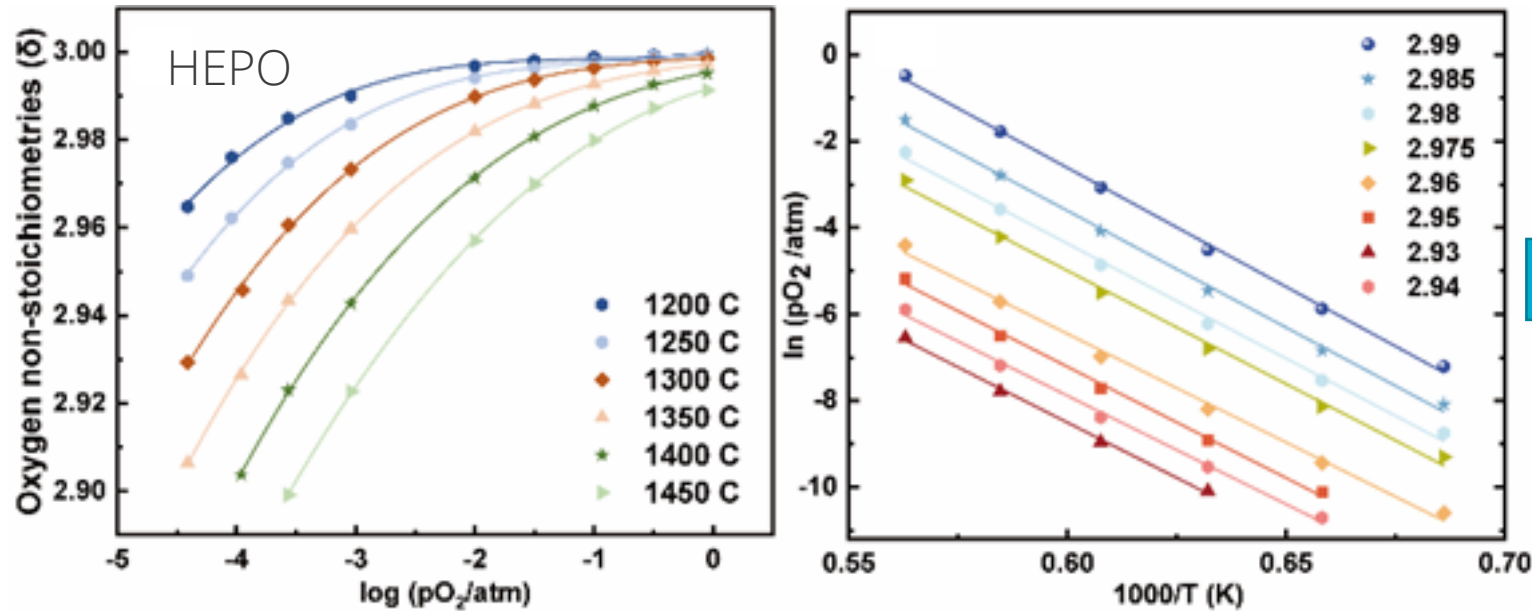


Shale Oil and Gas Handbook (2015)

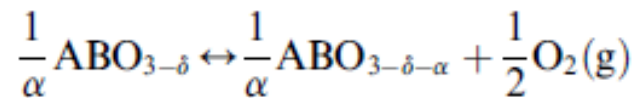


\*In collaboration with X. Liu (WVU) and J. Luo (UCSD)

# H<sub>r</sub> Decrease for HEPO



Van't Hoff analysis



$$K_{\text{redox}} = (pO_2^*)^{1/2} = \exp\left(-\frac{\Delta H_{\text{redox}}^0 - T\Delta S_{\text{redox}}^0}{RT}\right)$$

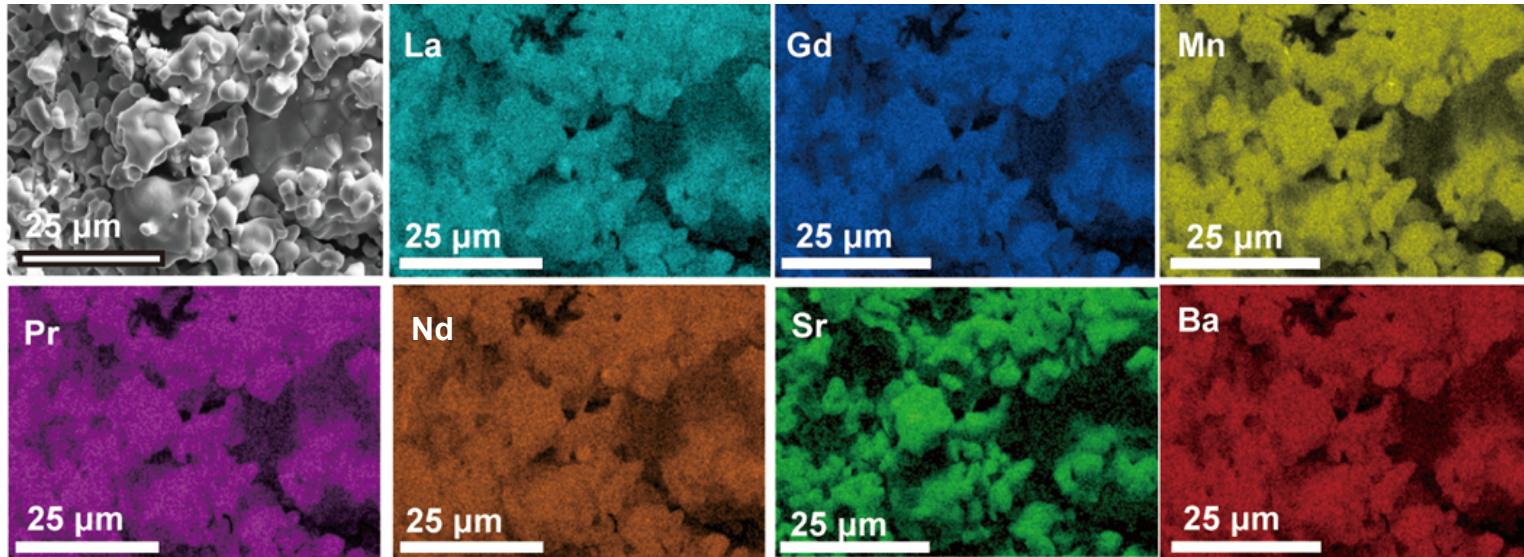
Takacs, et al., Acta Mater., 103 (2016) 700  
Mizusaki, et al., Solid State Ionics, 129 (2000) 163  
Yang, J. Mater. Chem. A, 2 (2014) 13612

- Lower  $H_r$  for HEPO compared to LSM
- Entropy similar

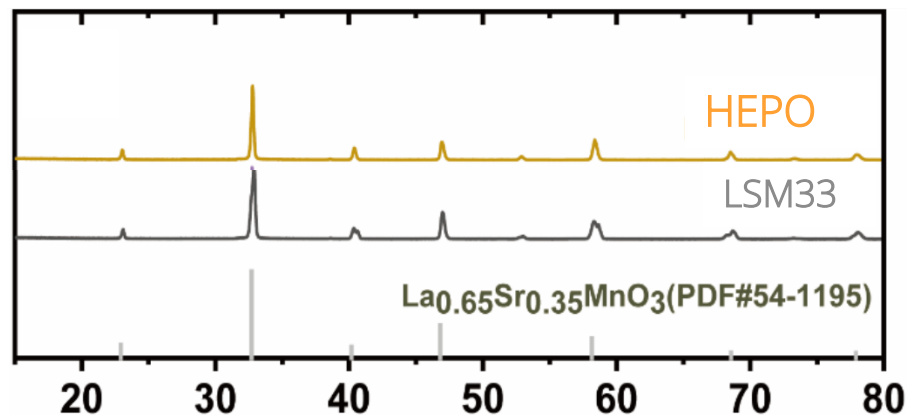
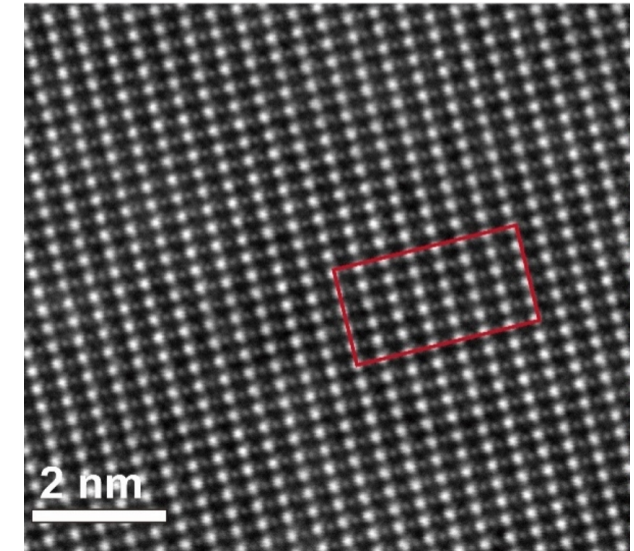


# Homogeneous Composition and Disordered Atomic Structure

SEM/EDS: No significant evidence of compositional heterogeneity



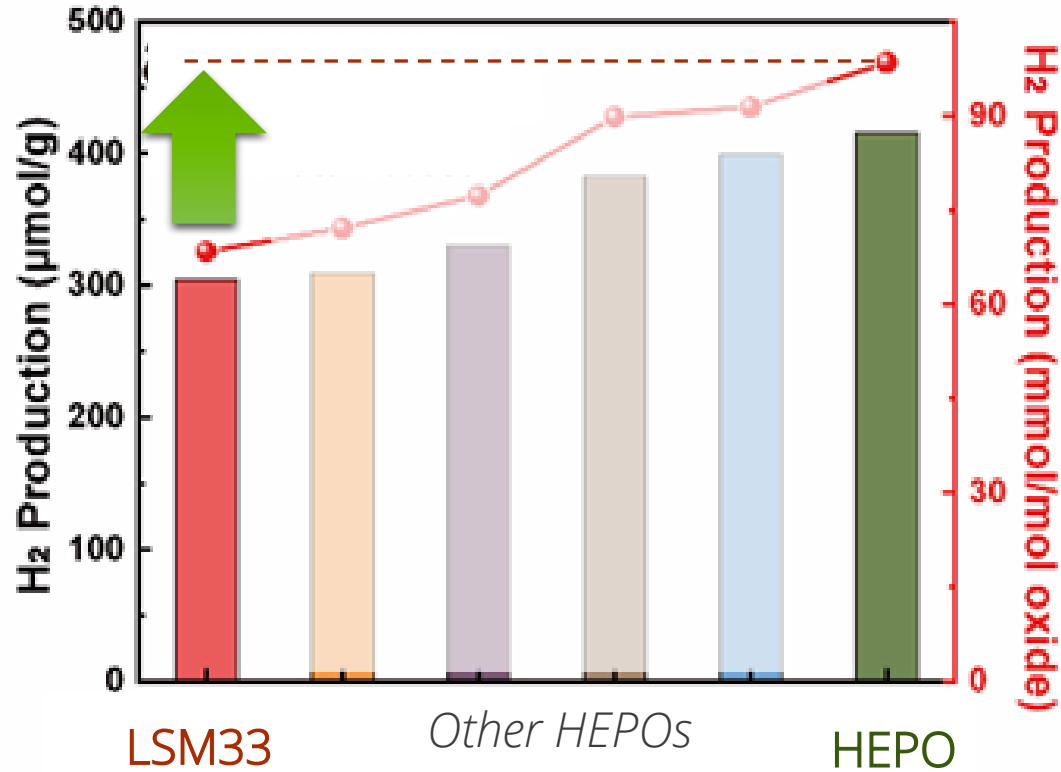
HAADF-STEM: No significant evidence of A-site cation ordering



- Expected perovskite structure of HEPO and LSM33
- HEPO ~0.2% smaller cell volume



# Dramatic Increase in STCH H<sub>2</sub> Production



46% increase in H<sub>2</sub> production with HEPO compared to LSM33!

Cijie Liu et al., J. Mater. Chem. A, 2024,12, 3910-3922

1350 °C N<sub>2</sub> reduction; 1100 °C 40% steam oxidation

# Summary

- Presented exemplar STCH materials and key STCH metrics
  - Identified significant gap in  $H_2$  production in low steam/ $H_2$  ratio materials
- Demonstrated successful water splitting with ML predicted compound
  - Improved thermal stability with Al addition
  - Competitive  $H_2$  production to  $CeO_2$  in low steam/ $H_2$
  - Despite severe morphological and phase changes, maintains  $H_2$  production after many cycles
- Examined high temperature ( $>1200\text{ }^\circ\text{C}$ ) oxygen non-stoichiometry in LSM20
  - Experimental differences from thermodynamic predictions in high  $pO_2$
- Discussed new  $(La_{1/6}Pr_{1/6}Nd_{1/6}Gd_{1/6}Ba_{1/6}Sr_{1/6})MnO_{3-\delta}$  (HEPO) STCH material with much lower reduction enthalpy than LSM of same acceptor content

# Acknowledgements

- U.S. Department of Energy (DOE)/ Hydrogen and Fuel Cell Technologies Office (HFTO) HydroGEN Advanced Water Splitting Materials Consortium ([h2awsm.org](http://h2awsm.org))
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