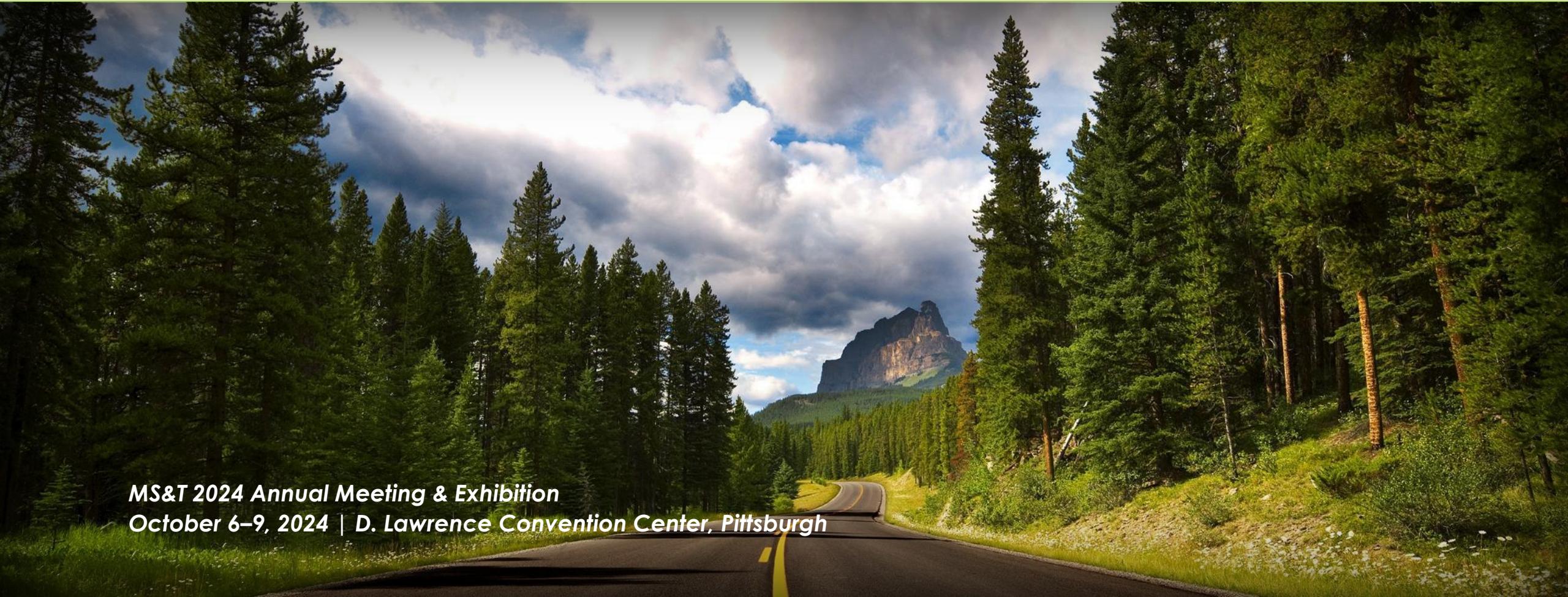


Revolutionizing Materials Design: The Intersection of Quantum Mechanics and Data Modeling



Wissam A. Saidi

*Computational Materials Engineering Team, Head
National Energy Technology Laboratory*



MS&T 2024 Annual Meeting & Exhibition

October 6–9, 2024 | D. Lawrence Convention Center, Pittsburgh

Saidi Lab

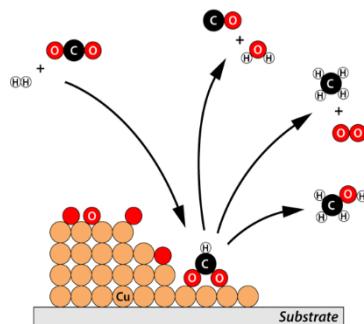
Department Chair



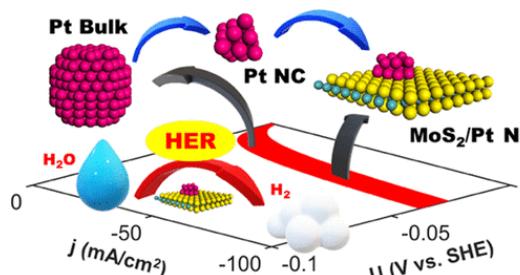
Department of Mechanical Engineering and
Materials Science
University of Pittsburgh



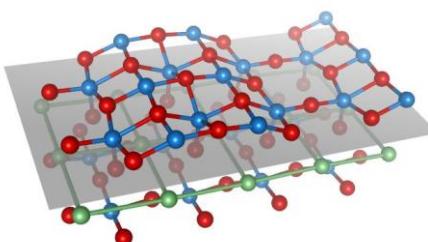
Oxidation and
Corrosion



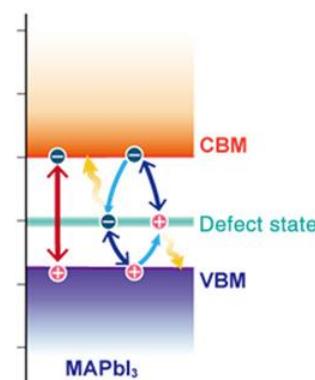
Catalysis



Electrocatalysts



Nano Materials



Solar cells

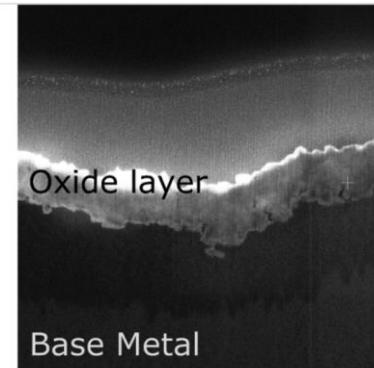
Industrial & Engineering Chemistry Research > Vol 58/Issue 3 > Article

Cite Share Jump to Expand

ARTICLE | December 24, 2018

Mechanism behind the Inhibiting Effect of CO₂ on the Oxidation of Al-Mg Alloys

Nicholas Smith*, Brian Gleeson, Wissam A. Saidi, Anne Kvithyld, and Gabriella Tranell



Effects of CO₂ Cover Gas and Yttrium Additions on the Oxidation of AlMg Alloys

Conference paper | First Online: 16 February 2019

pp 1025–1032 | [Cite this conference paper](#)

N. Smith, B. Gleeson, W. Saidi, A. Kvithyld & G. Tranell

Materials Design for energy technologies through
multiscale simulations

National Energy Technology Laboratory (NETL)



One of 17 U.S. Department of Energy (DOE) national laboratories; producing technological solutions to America's energy challenges.

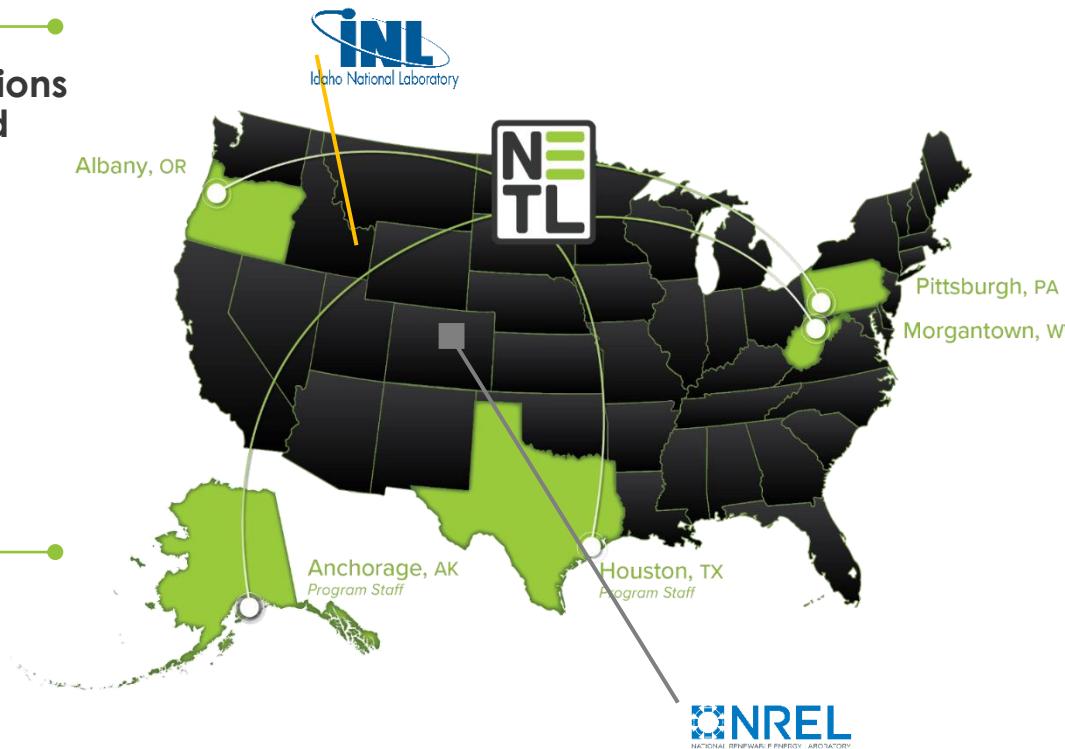
MISSION

Driving innovation and delivering solutions for an environmentally sustainable and prosperous energy future:

- Ensuring affordable, abundant and reliable energy that drives a robust economy and national security, while
- Developing technologies to manage carbon across the full life cycle, and
- Enabling environmental sustainability for all Americans.

VISION

To be the nation's premier energy technology laboratory, delivering integrated solutions to enable transformation to a sustainable energy future.

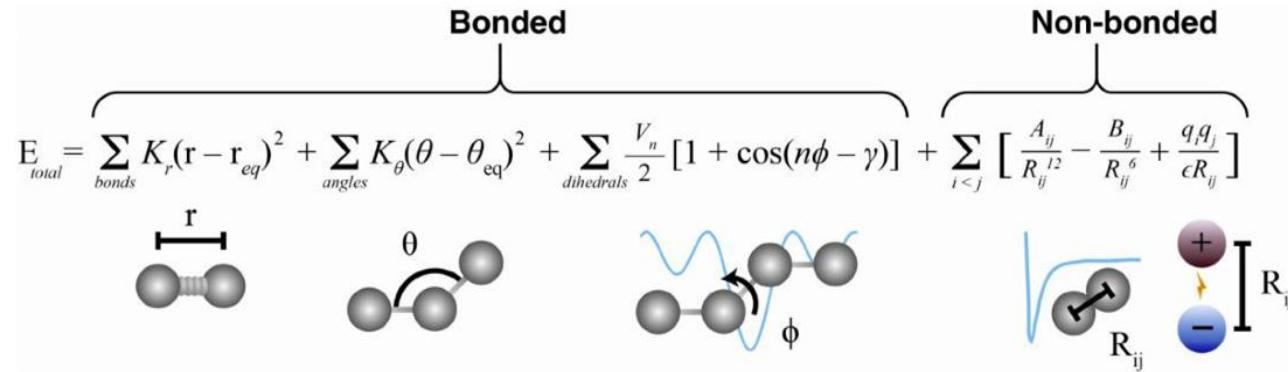


- NETL has **three** research laboratories
- **One of three** applied research national labs
- Government owned & operated
- Leader in cutting-edge research in CO₂ conversion to higher-value products
- Only National Lab dedicated to carbon management research

Atomistic Force-Fields Indispensable Tool for Materials Design



Traditional Approach



Limitations of traditional forcefields:

- **Time-intensive (low training transparency)**
- **Limited accuracy and transferability**
- **Unavailable for many materials**

AI/ML

Big data



**ML atomistic
potentials accelerate
DFT without
comprising accuracy**

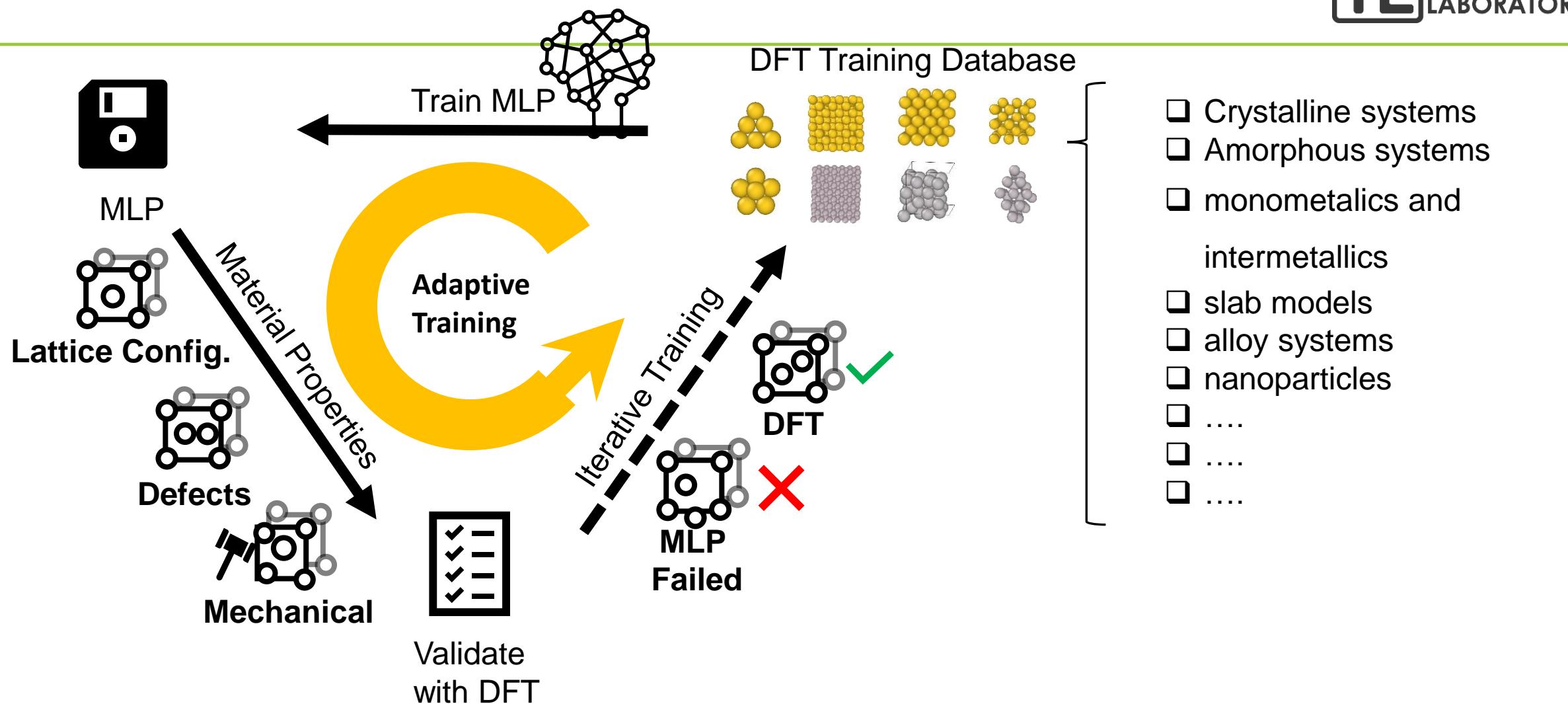
- ❖ Melting
- ❖ Oxidation of binary alloy system

Noble prize 2024 in physics:
Hopfield and Hinton

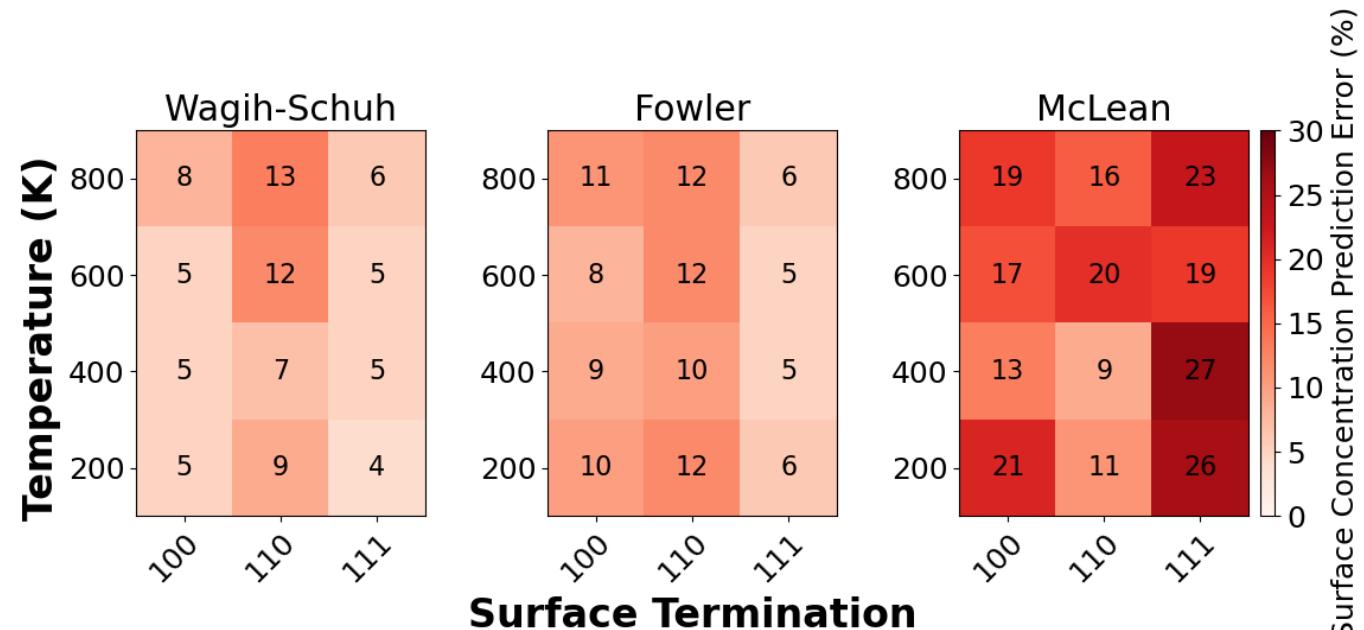
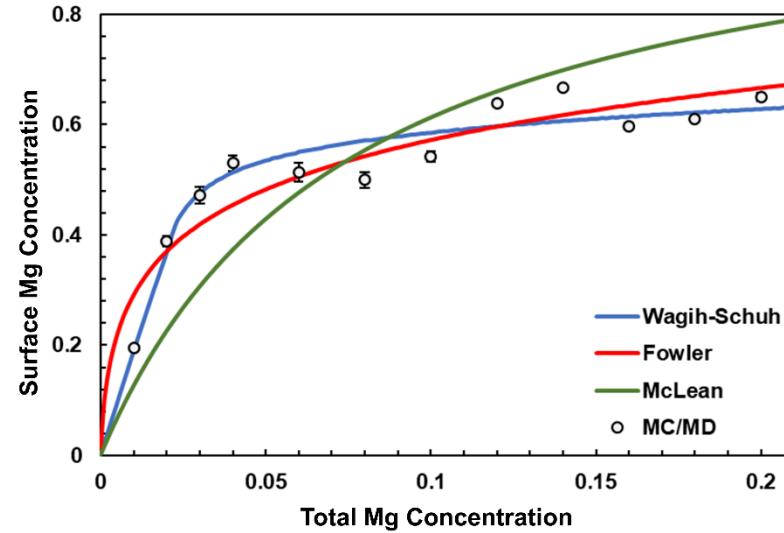


U.S. DEPARTMENT OF
ENERGY

MLP Workflow: Training and Validation



Temperature Dependent Mg Surface Segregation AlMg Alloys: Isotherm Comparison

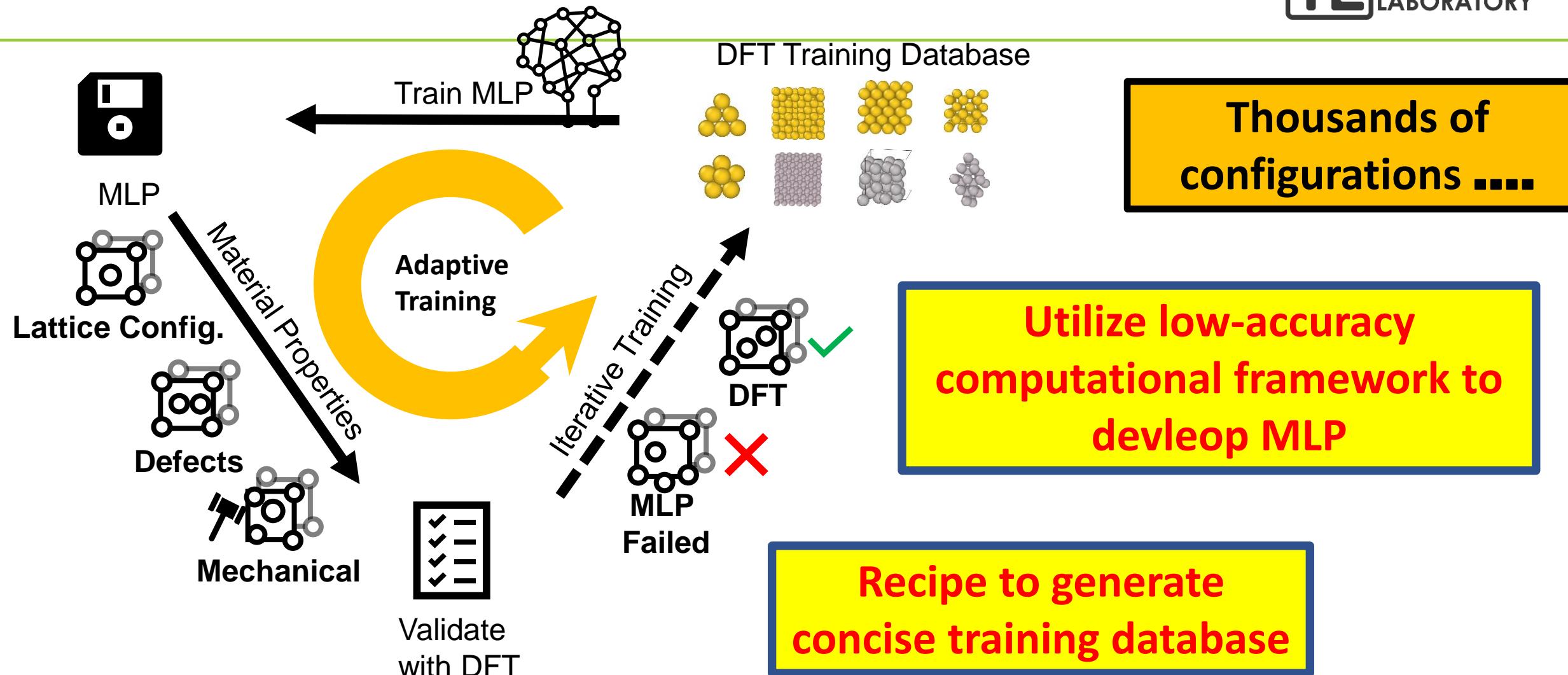


Wagih-Schuh Isotherm (Modified for Surfaces)

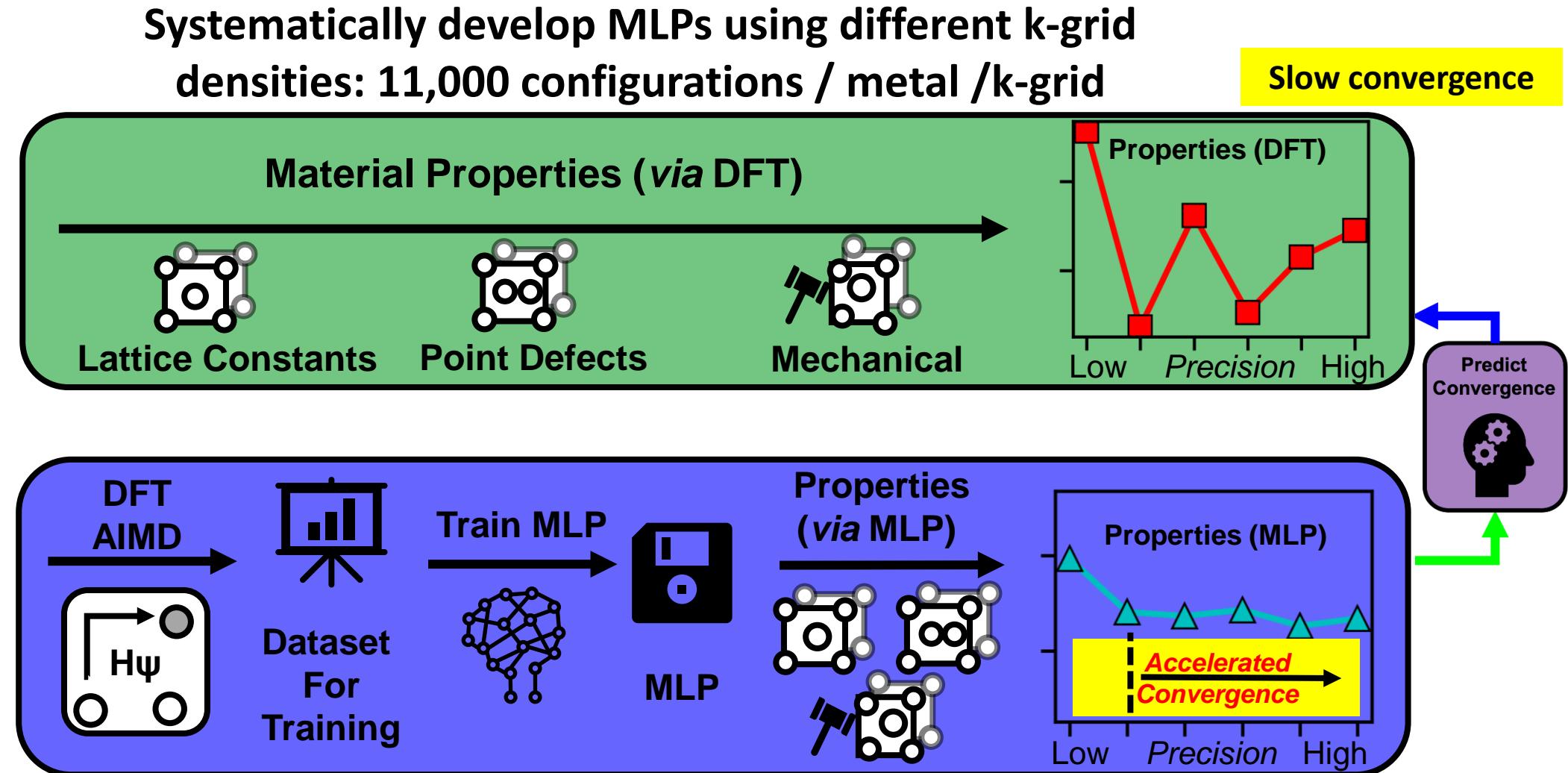
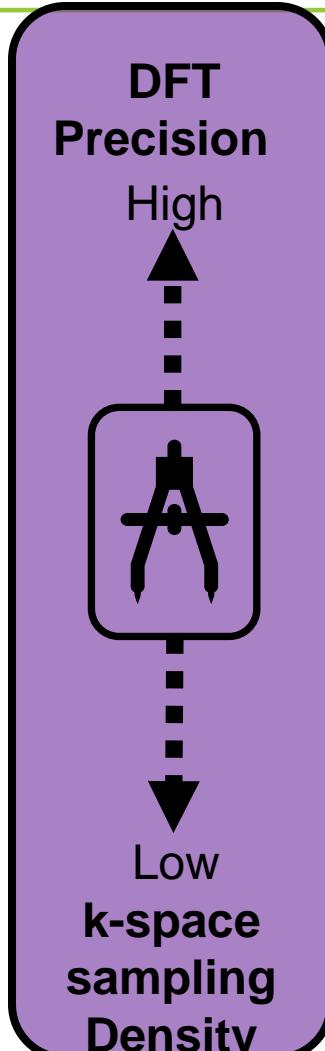
$$\Delta E^\omega(X_{surf}) = \begin{cases} 0 & X_{surf} \leq X_0 \\ \omega(X_{surf} - X_0) & X_{surf} > X_0 \end{cases} \quad X_{tot} = (1 - f_{surf}) X_{bulk} + f_{surf} (A)^{-1} \quad A = \left[1 + \frac{1 - X_{bulk}}{X_{bulk}} \cdot \exp\left(\frac{\Delta \bar{E}_{seg} + \Delta E^\omega}{kT}\right) \right]^{-1}$$

MLP results match most precisely with Wagih-Schuh Isotherm

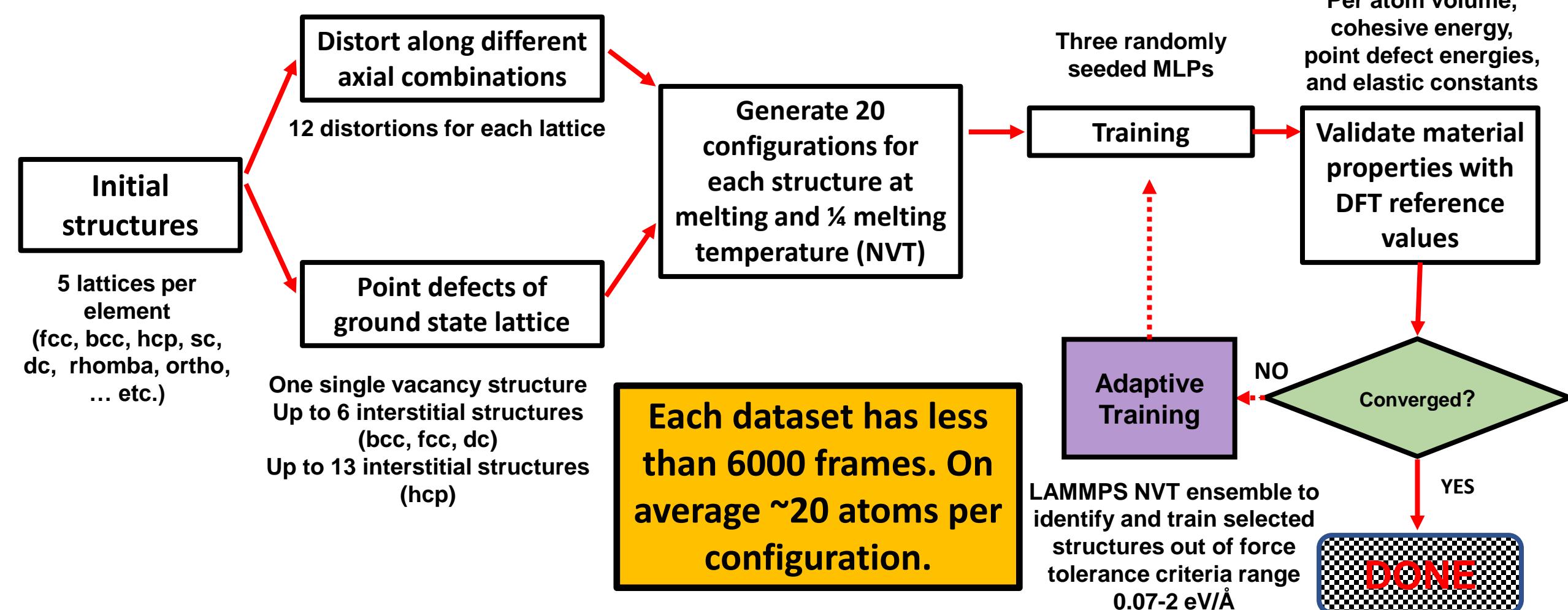
MLP Workflow: Training and Validation



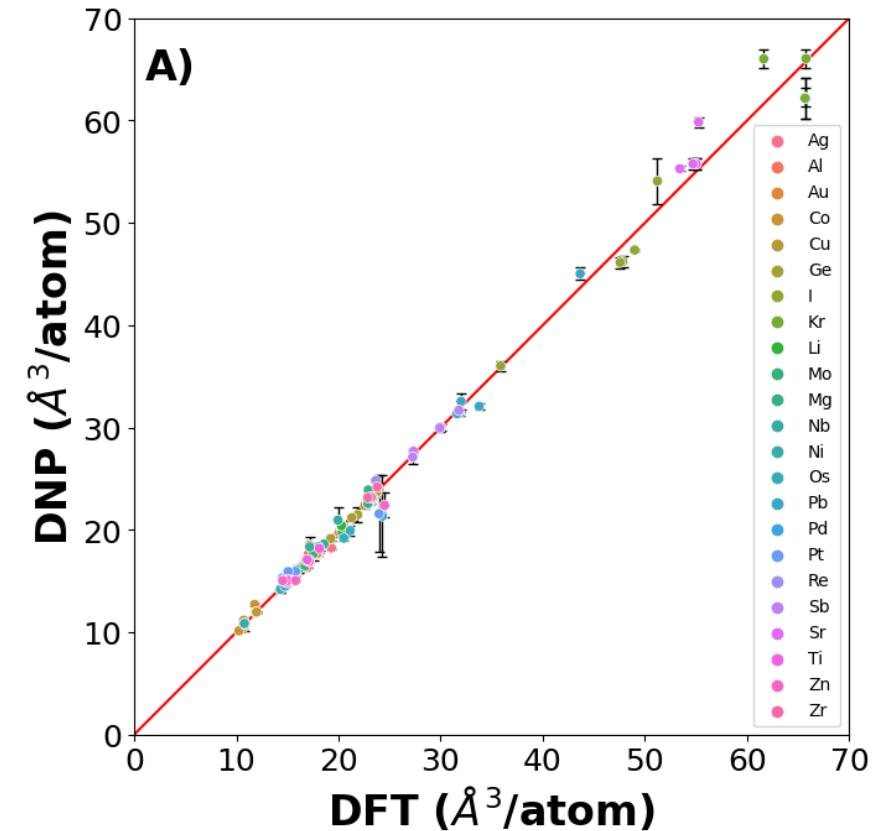
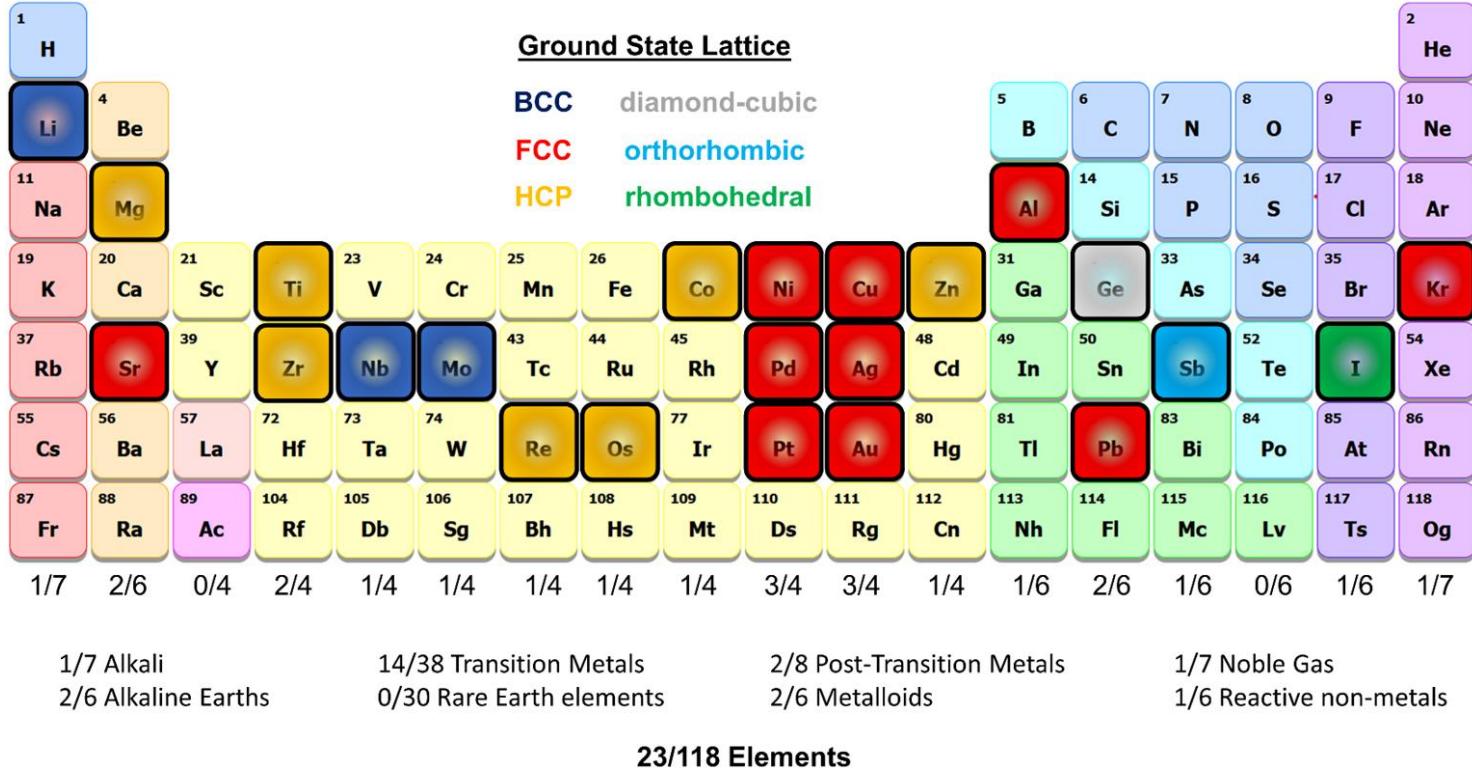
Computational Experiments on Al, Cu and Mg



Recipe for concise dataset....



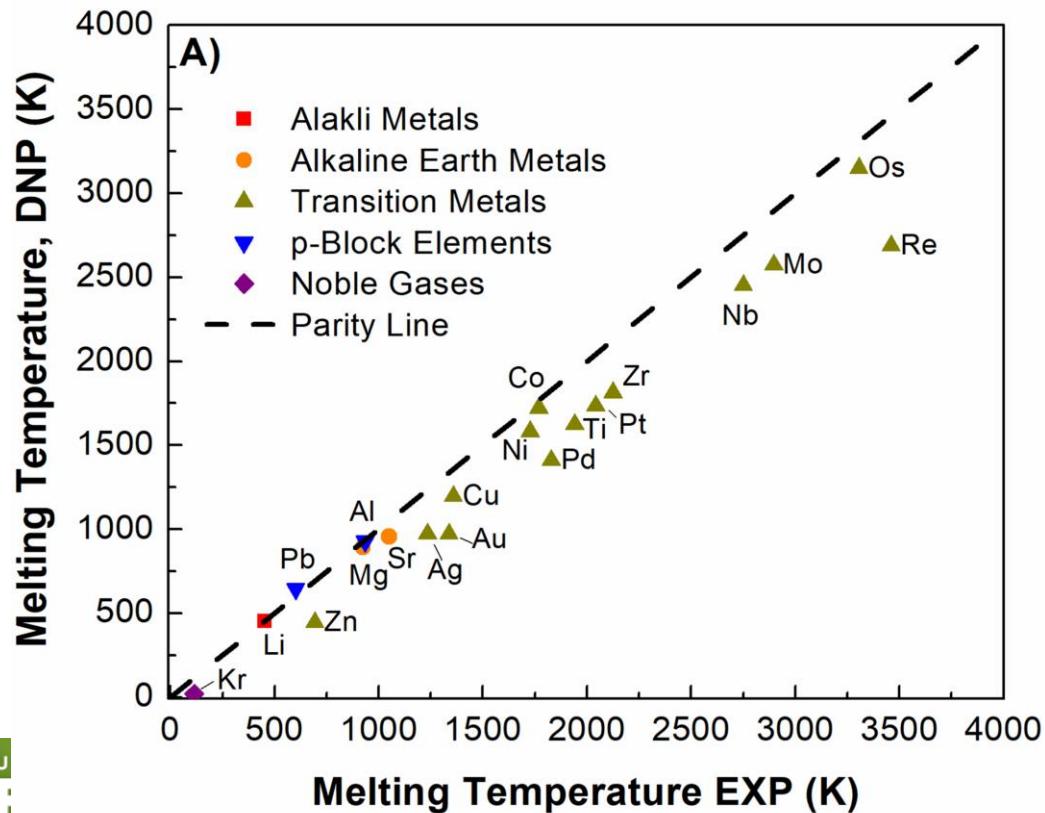
Single Element MLPs



Understanding Material Melting

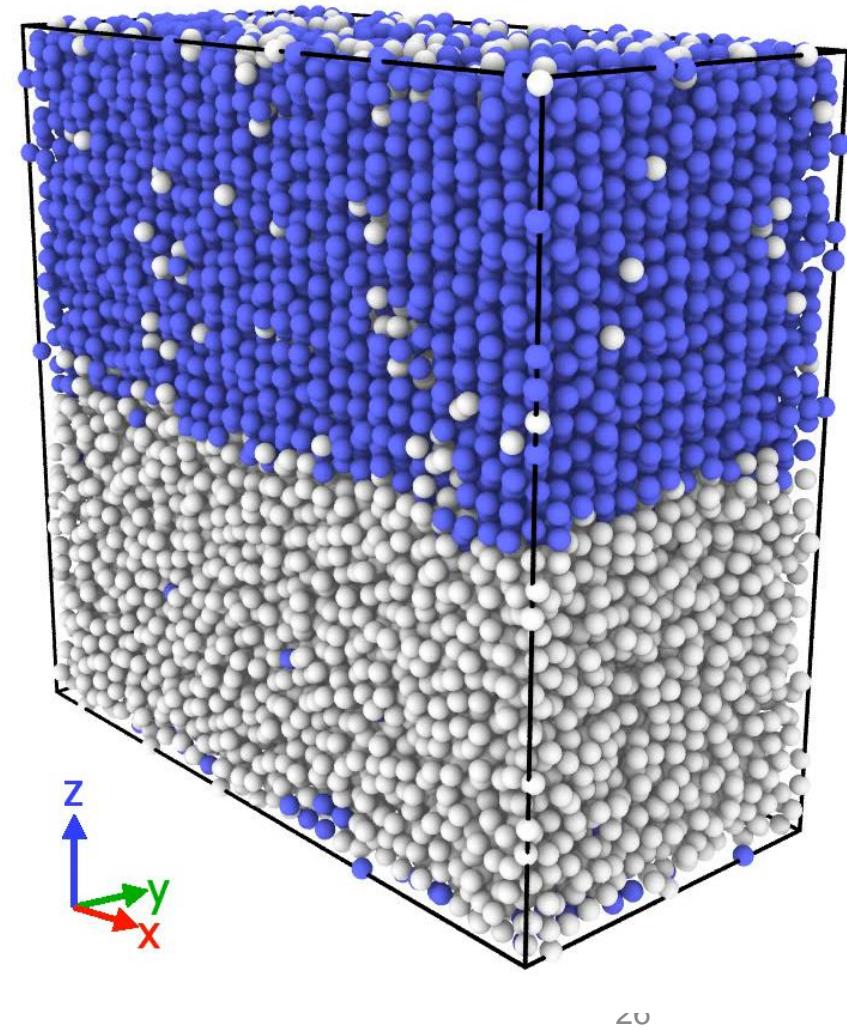
- ❖ Two-Phase Coexistence (TPC) approach is the “gold-standard” **but requires system sizes in excess of 10,000 atoms!**

Applied MLP to compute T_m of 20 elements

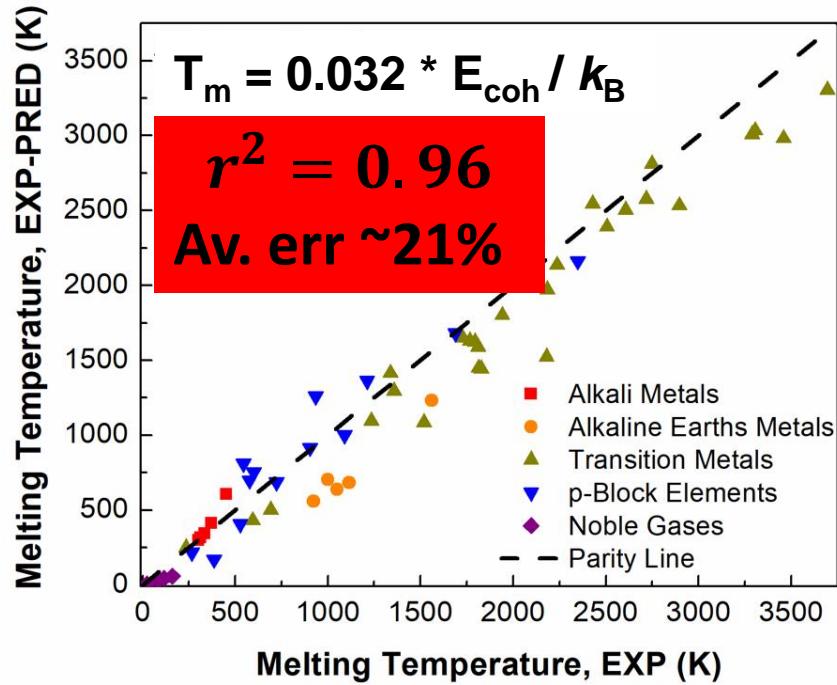


Mo
BCC

Liquid



Estimate T_m from Cohesive Energy

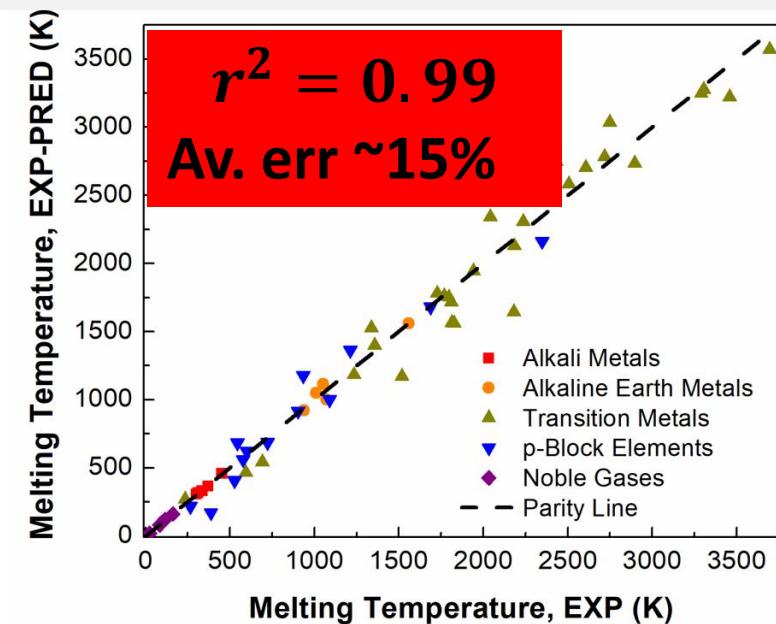


Using group-specific trends

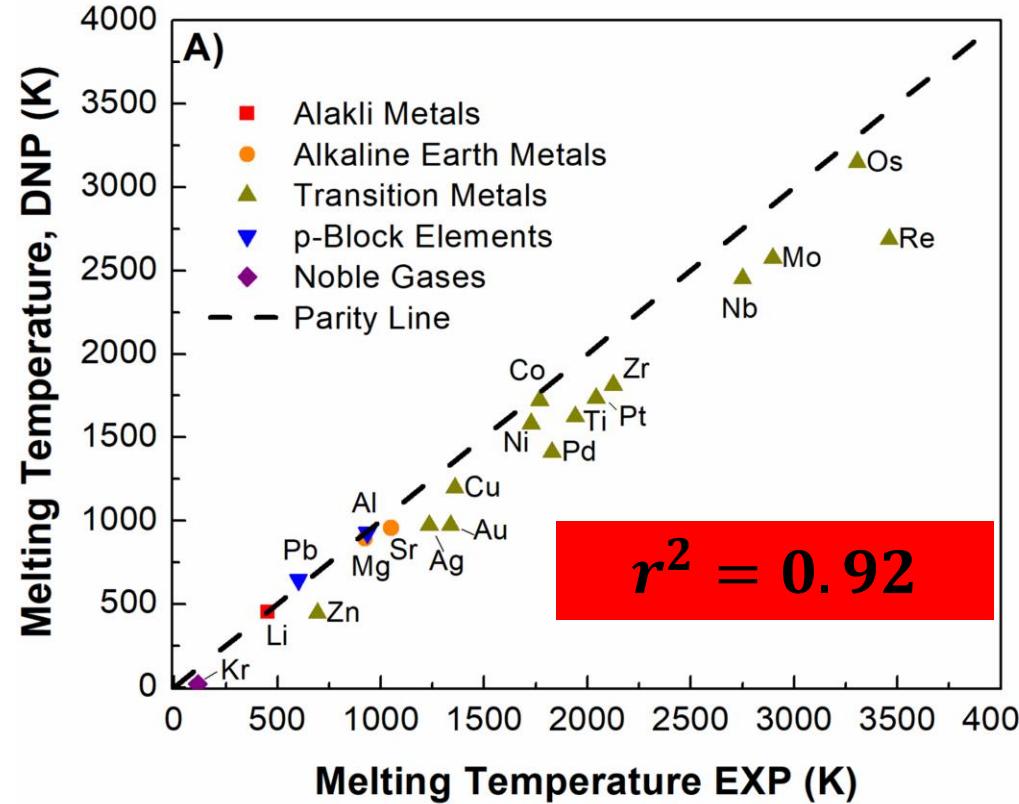


The “Universal” equation can be improved by sorting elements by periodic group

$T_m = (340 \text{ K/eV} * E_{coh}) + 430 \text{ K}$ (Alkali Metals)
 $T_m = (181 \text{ K/eV} * E_{coh}) + 162 \text{ K}$ (Alkaline Earths)
 $T_m = 401 \text{ K/eV} * E_{coh}$ (Transition Metals)
 $T_m = (407 \text{ K/eV} * E_{coh}) - 204 \text{ K}$ (p-block)
 $T_m = 1012 \text{ K/eV} * E_{coh}$ (Noble Gases)



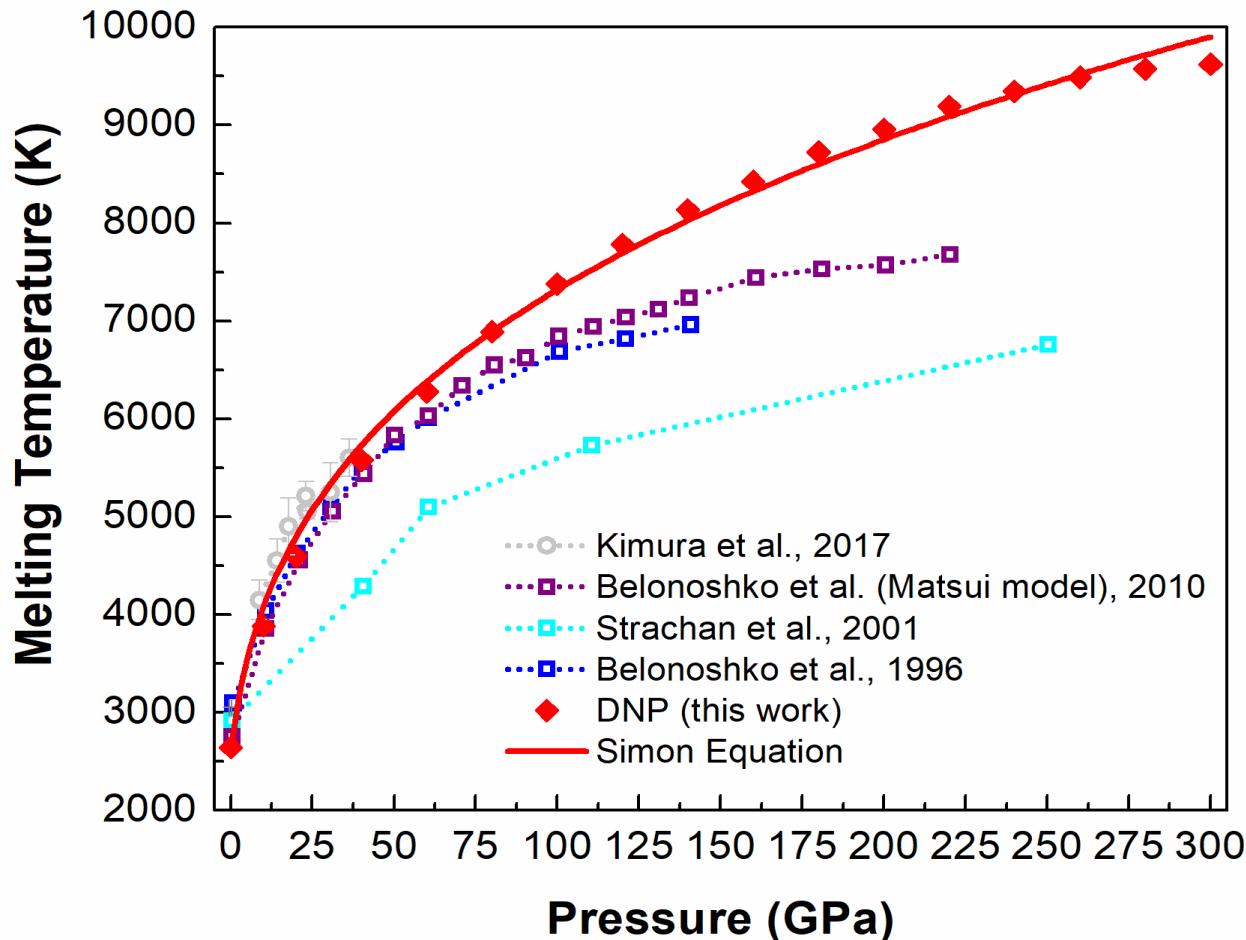
Comparing TPC-MLP to Experiment and DFT-Predicted



DFT/MLP underestimates experimental T_m

Improved agreement MLP and DFT-predicted.

MgO Solid-Liquid Phase Transitions at Mantle Conditions

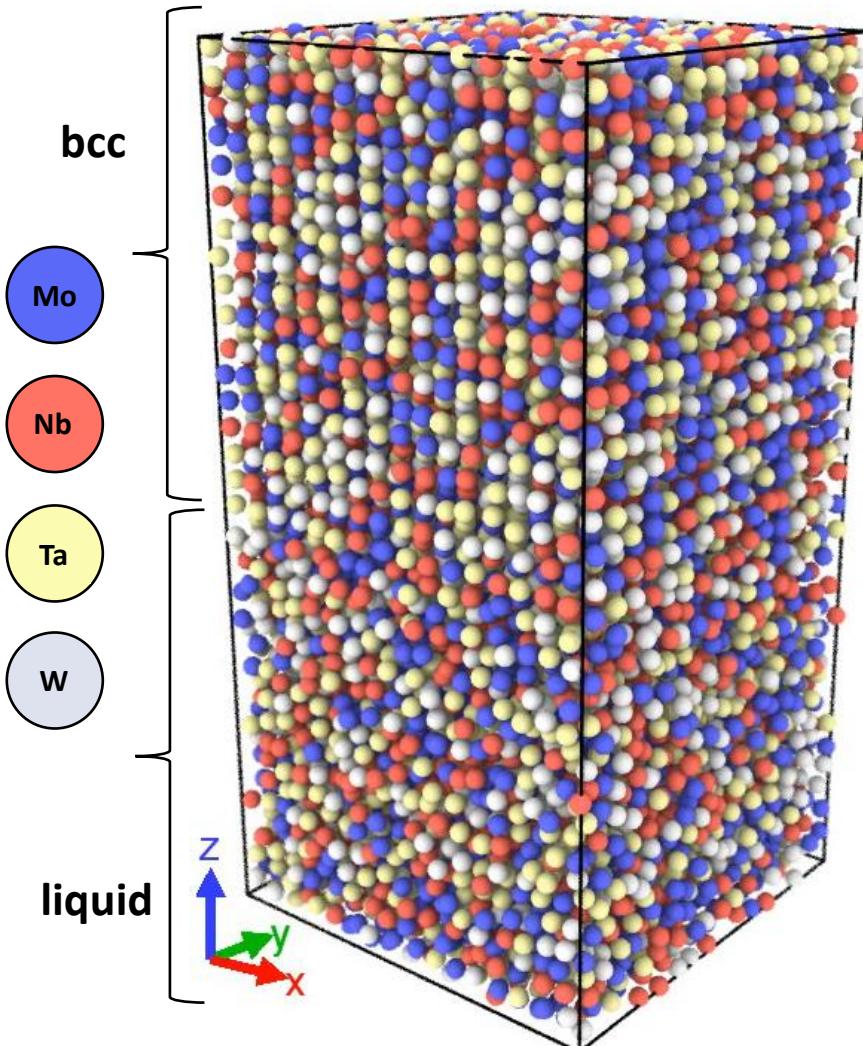


We can describe solid-liquid phase changes for large supercells over long time scales and extreme pressures ~ 300 GPa with DFT accuracy

Preliminary Results: MoNbTaW HEA



MD MLP-TPC



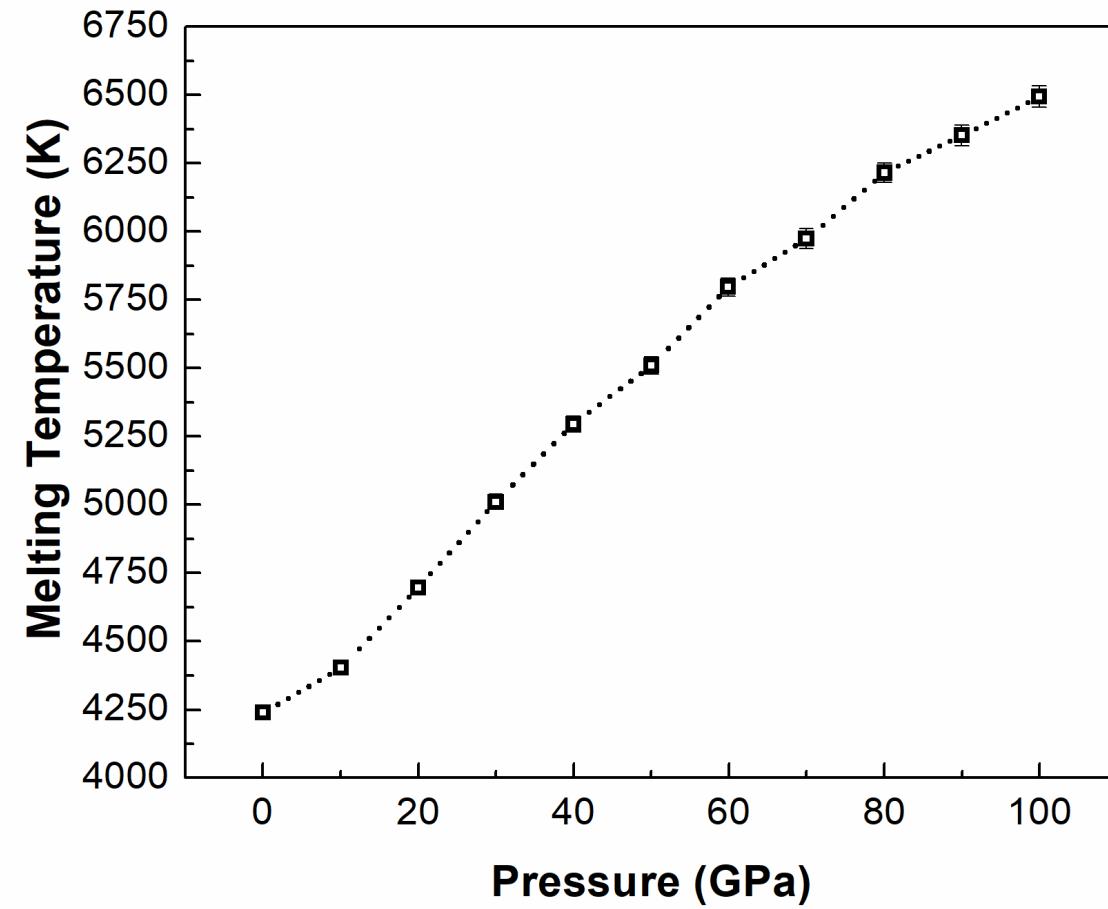
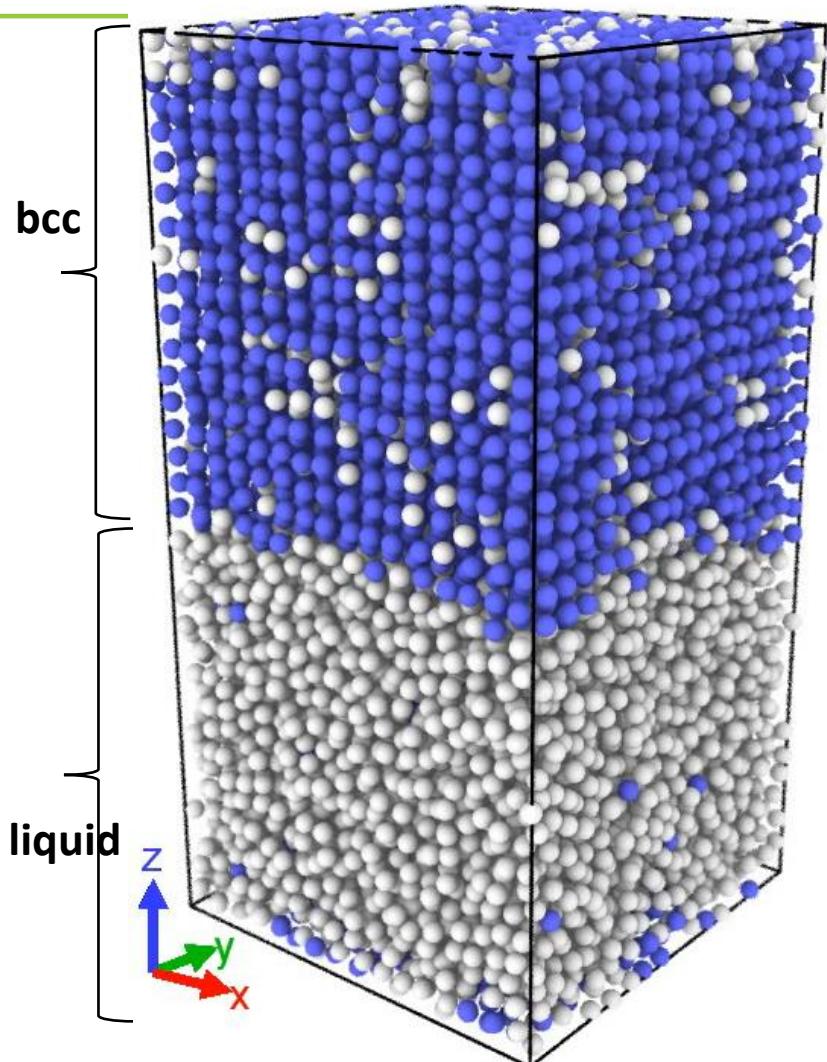
MoNbTaW Composition	E_{coh} (eV)	T_m -PRED (K)	T_m TPC-MLP (K)
Mo256.Nb256.Ta256.W256-r1	-10.211	4100	4216 \pm 28
Mo256.Nb256.Ta256.W256-r2	-10.210	4099	4221 \pm 21
Mo256.Nb256.Ta256.W256-r3	-10.208	4099	4217 \pm 22
Mo256.Nb256.Ta256.W256-r4	-10.210	4100	4220 \pm 21
Mo128.Nb384.Ta128.W384	-10.153	4077	4267 \pm 22
Mo128.Nb128.Ta384.W384	-9.837	3950	4251 \pm 20
Mo128.Nb384.Ta384.W128	-10.229	4107	4241 \pm 22
Mo384.Nb128.Ta128.W384	-9.477	3805	4117 \pm 23
Mo384.Nb128.Ta384.W128	-10.154	4077	4320 \pm 19
Mo384.Nb384.Ta128.W128	-9.622	3864	4215 \pm 28

- Similar T_m over narrow compositional changes.
- Predicted T_m from E_{coh} agrees well with MLP-TPC T_m



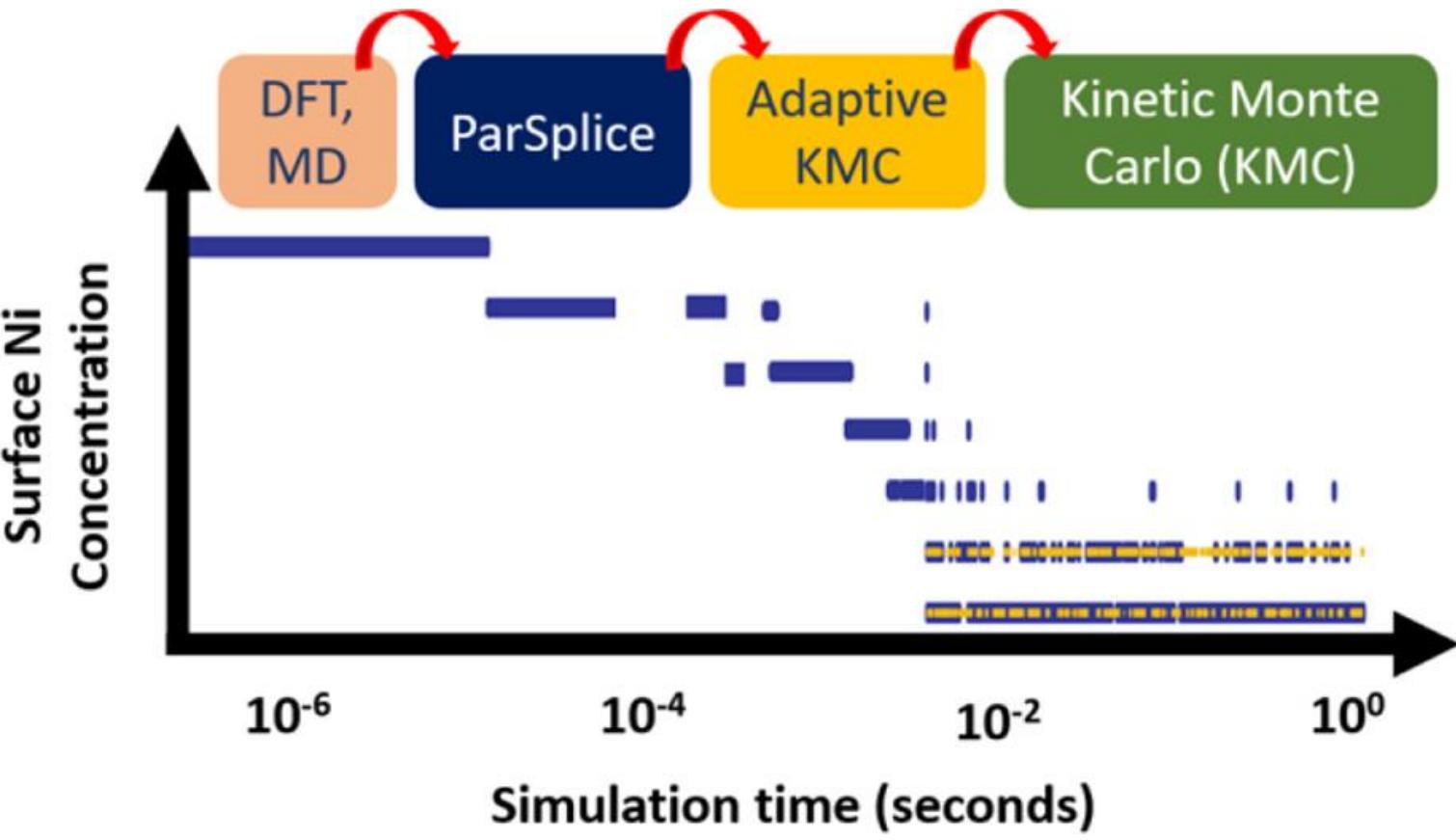
Preliminary Results: MoNbTaW HEA

MD MLP-TPC



MLP describes T_m behavior well at elevated pressure

The trend for Cu-Ni Alloy Segregation

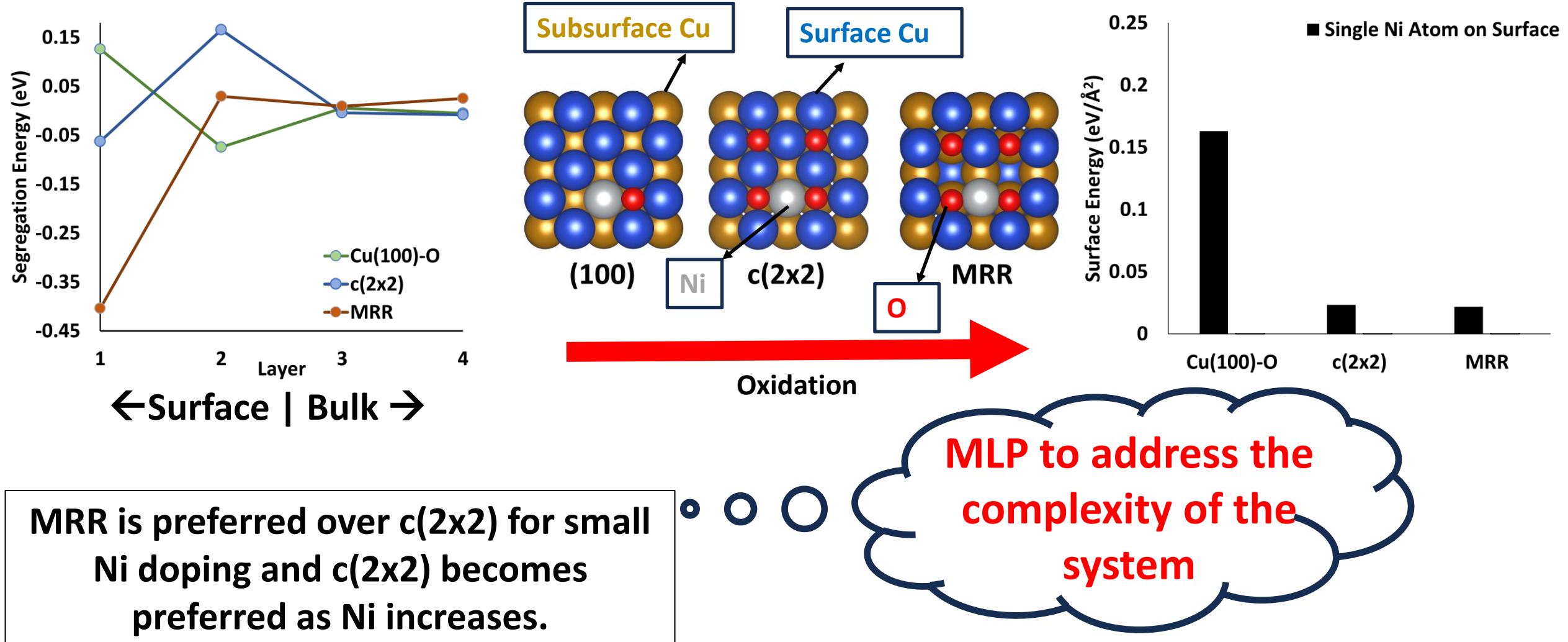


CuNi dynamics from ns → s
Cu segregates to the surface under reducing conditions

What happens under oxidizing conditions?

Competition between surface reconstructions and segregation tendencies

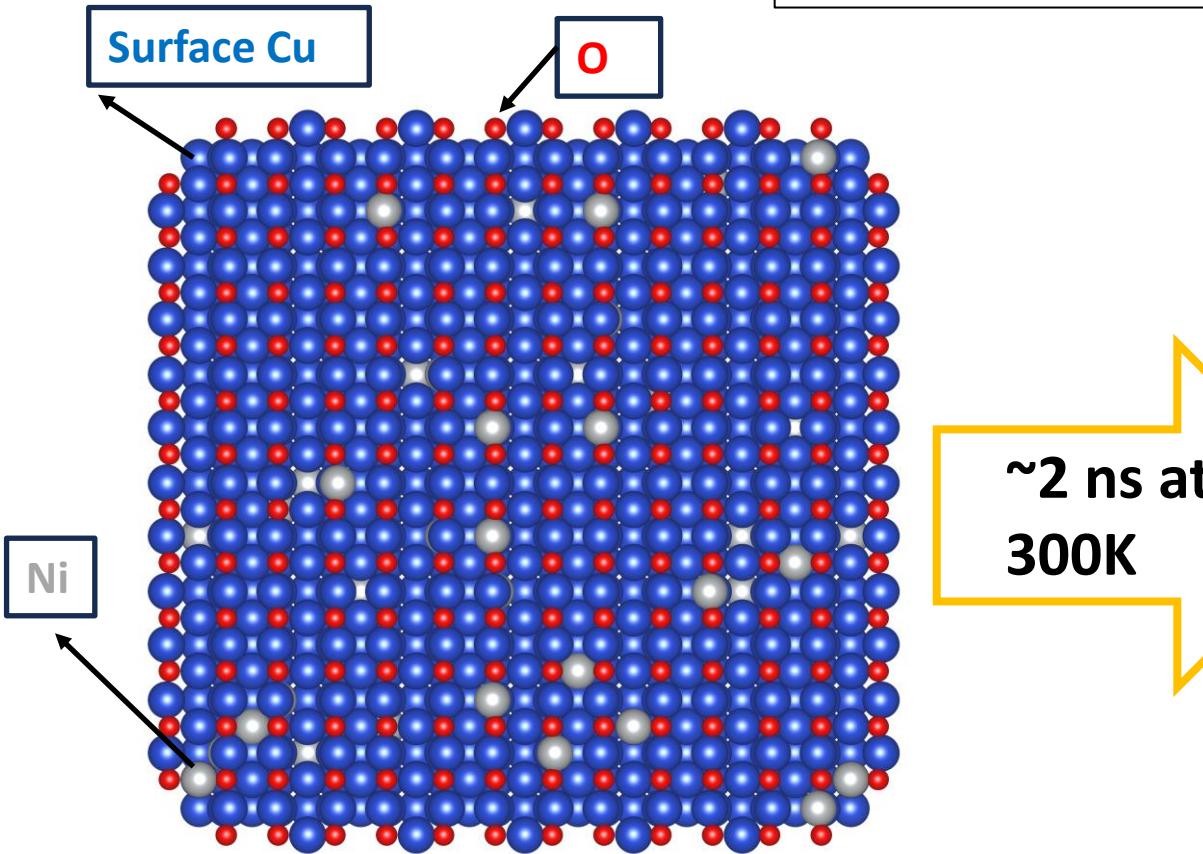
DFT segregation preferences on Cu (100) reconstructions:



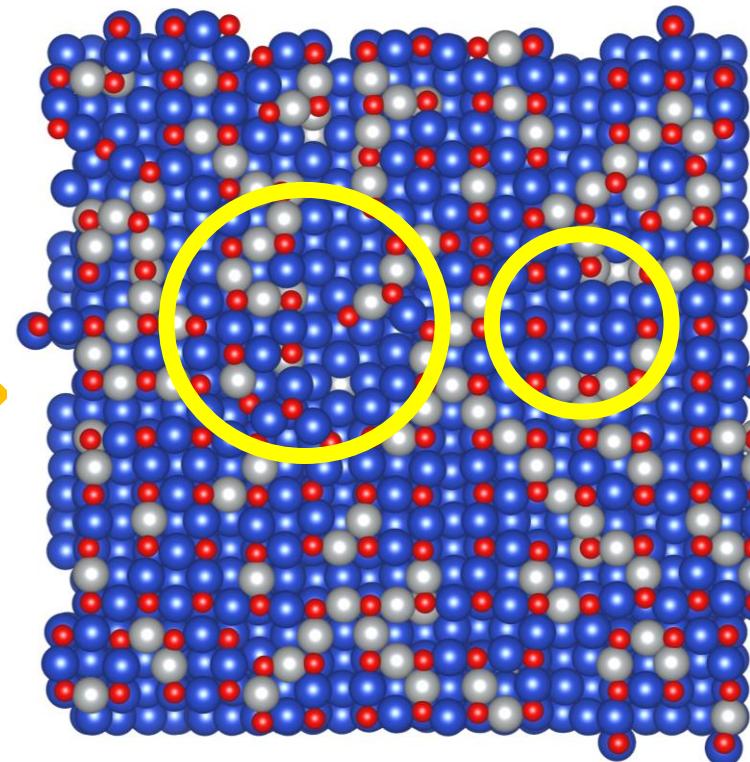
Segregation at finite time – MC/MD Simulation



MRR with random 5% Ni



~2 ns at
300K

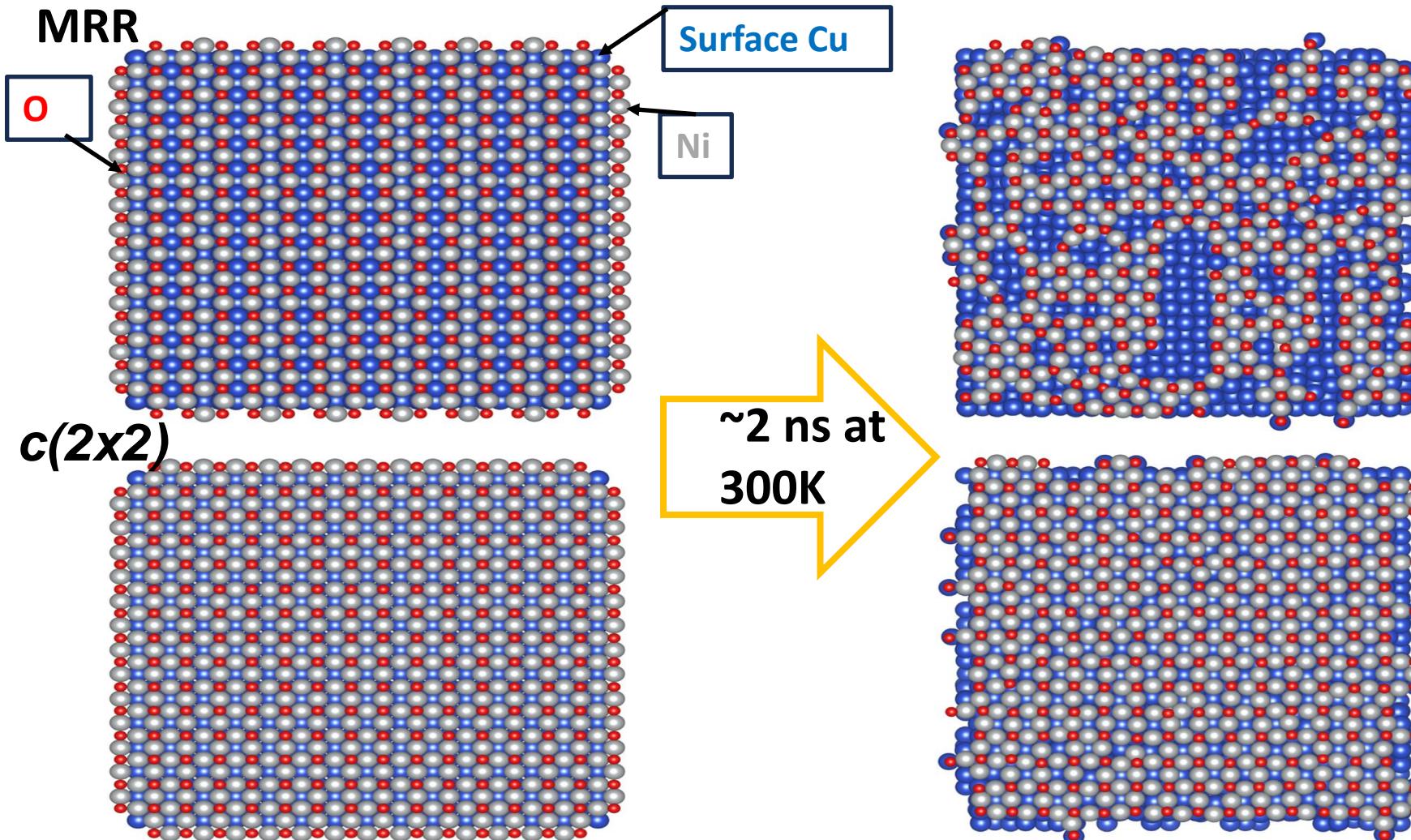


Ni segregation
and surface
rearrangement
to form Ni-O
bonds

Ni segregate to the surface and induce Ni-O nucleation

Simulations made
possible by MLP

Stability of full Ni monolayer ... MC/MD simulation

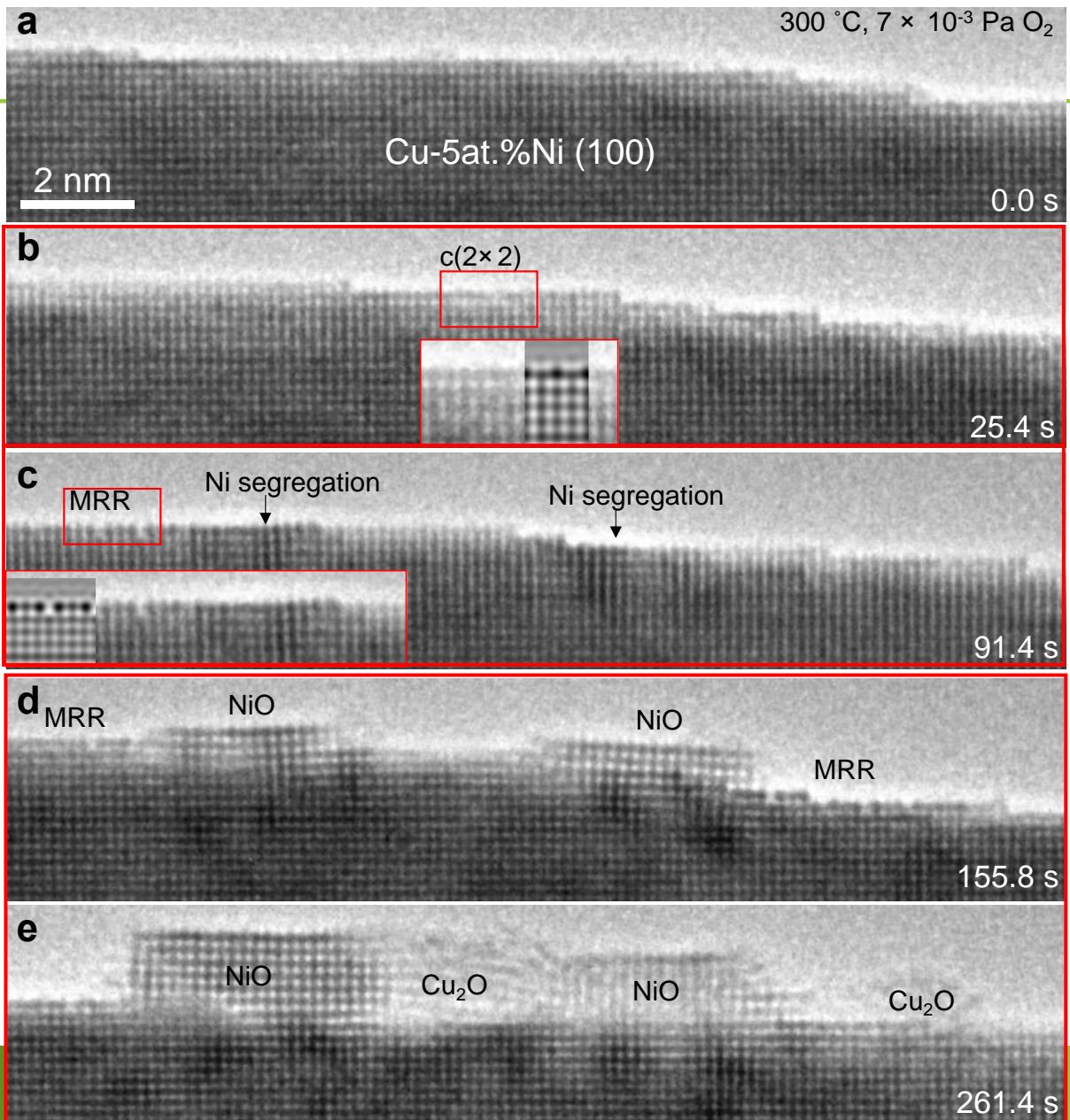


c(2x2) surface is more stable due to the NiO arrangement.

Simulations made possible by MLP

Experimental Verification

Collaboration with Li and Yang BNL



ETEM 5% Ni (300 °C, 7×10^{-3} Pa O₂)

- (b) shows the formation of c(2x2)
- (c) Ni segregates to the c(2x2), as highlighted by strain due to the difference between the lattice constant of Cu and Ni.
- At the same time MRR is forming from a region free of Ni.
- The c(2x2) with Ni eventually forms NiO, while the MRR forms Cu₂O.

DFT and MLP results are verified by experimental observation.

Accelerating Material Design for Energy Applications

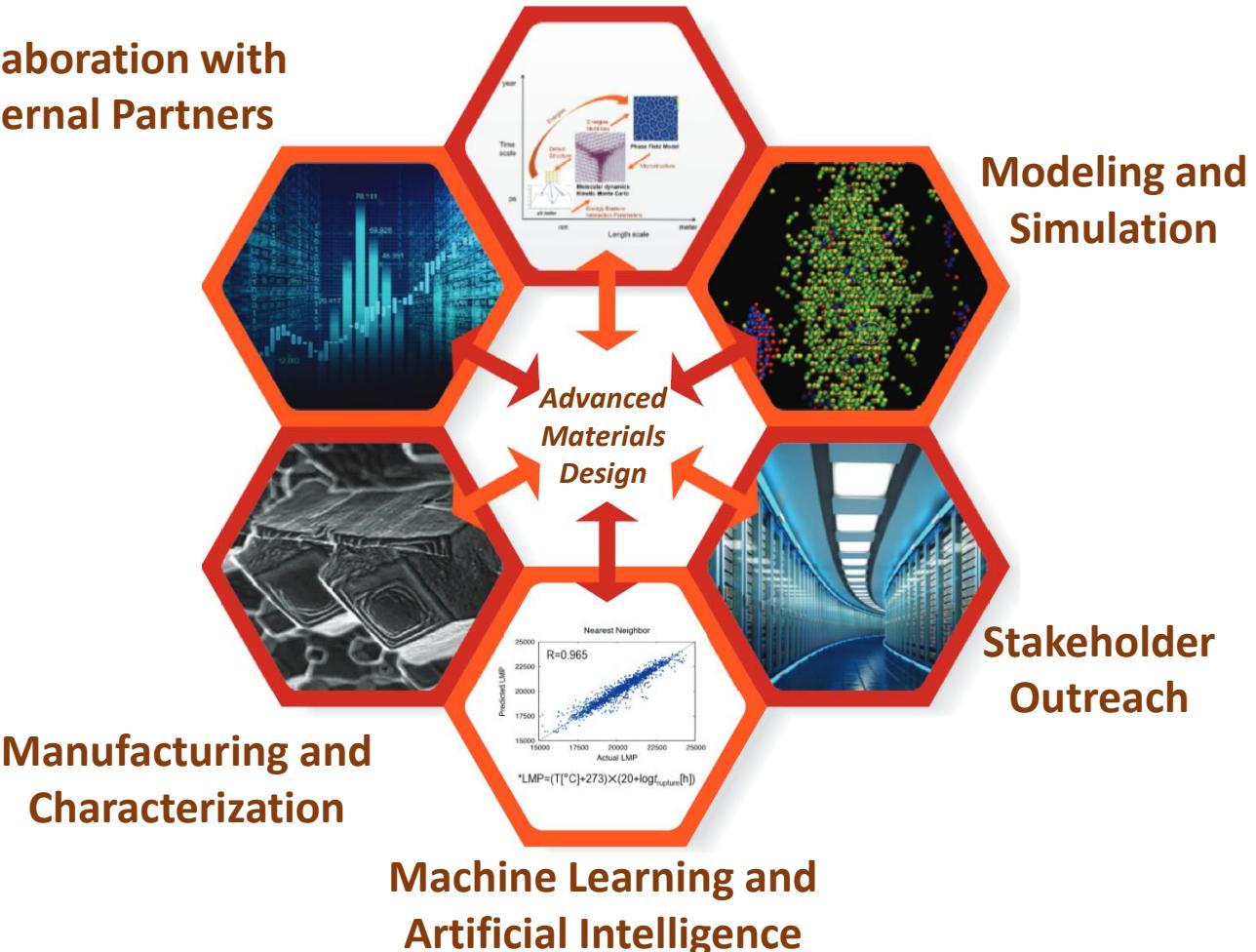
Integrating experimental, simulation and data-driven methods for material development



- **Accelerate development and deployment** of new materials for energy applications
- **Predict** long-term materials performance in fossil energy power plants of any cycle
- **Advance technologies** including High entropy alloys; structural alloys; hydrogen embrittlement; oxidation; catalysts; sorbents

Collaboration with External Partners

Experimental Validation

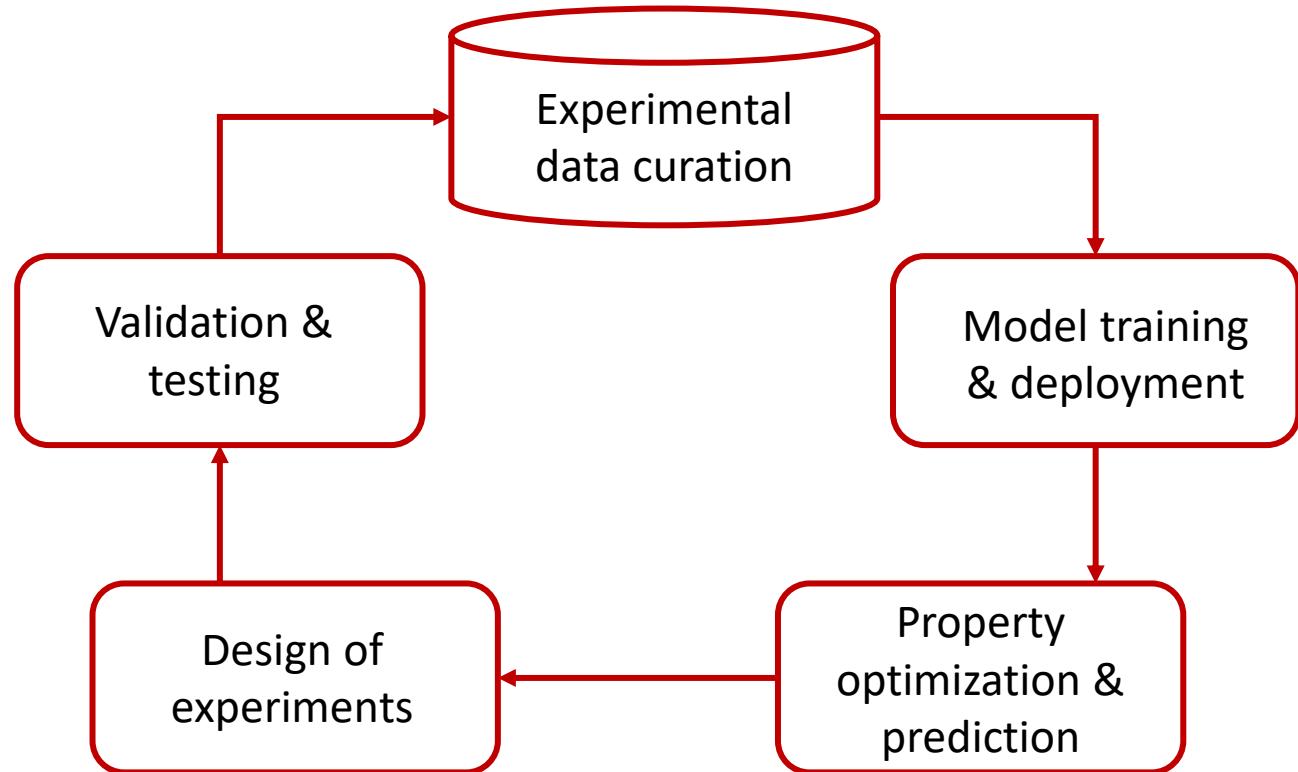


Integration into materials design & testing

Creep Rupture Time Prediction

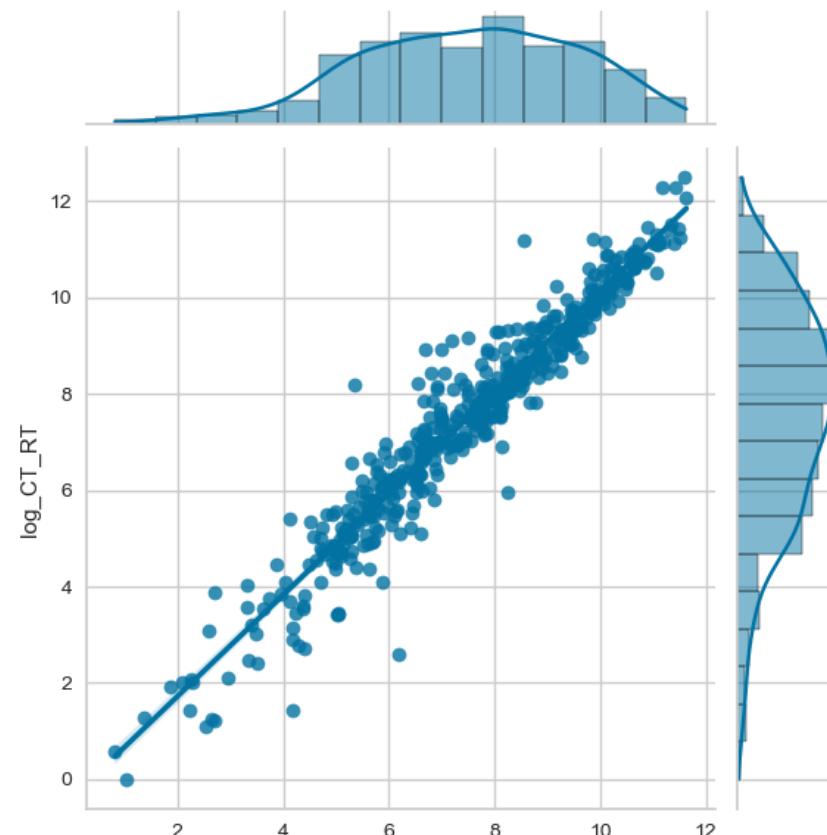


- Leveraging property prediction in material design loop
- Improving efficiency of experimentation through prediction of long-term properties
- Optimization of material performance



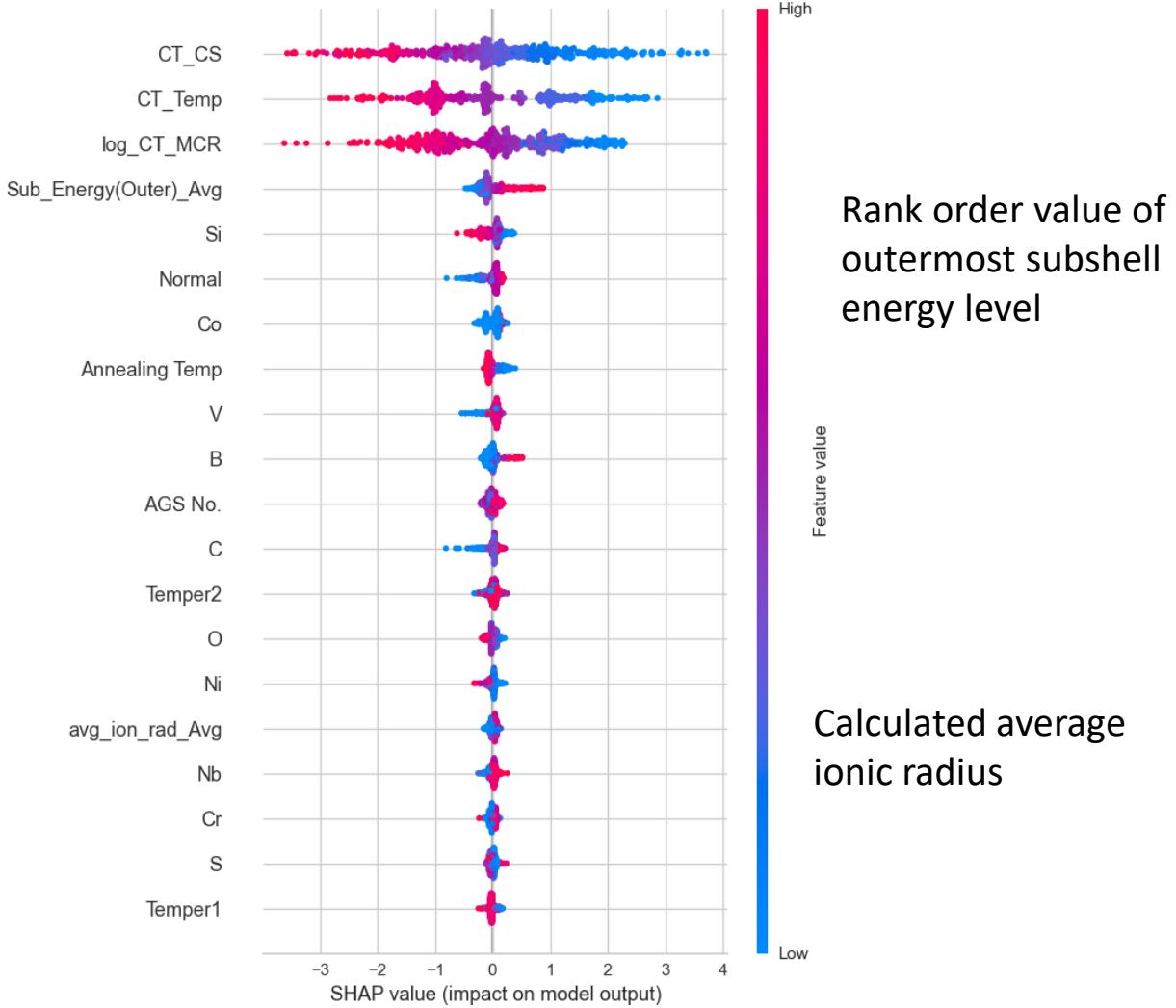
Creep rupture time prediction with physical features

CatBoost Regression of log Creep Rupture Time



Train $R^2: 0.9185 \pm 0.0051$

Hold-out test $R^2: 0.9282$



Addition of physical features

Aiding generalizability, interpretation

- Derived from Rule of Mixtures using at.% composition
- Adding physically relevant features related to mechanical behavior of alloys
 - Atomistic & physical properties
- Calculating 76 average and standard deviation values

Physical Properties	Atomistic Properties
Coefficient of linear thermal expansion (/K)	Number of electron vacancies
Electrical conductivity (10 ⁶ /cm Ω)	Number of valence electrons
Thermal conductivity (W/cmK)	Number of electrons in outermost subshell
Density (g/cc @ 300K)	Electron capacity of outermost subshell
Bulk modulus (Gpa)	Electron Vacancies in the outermost subshell
Rigidity Modulus (Gpa)	Rank order value of outermost subshell energy level
Youngs Modulus (Gpa)	Number of electrons in 2nd outermost subshell
Enthalpy of atomization (kJ/mole @ 25°C)	Electron capacity of 2nd outermost subshell
Enthalpy of fusion (kJ/mole)	Electrochemical Equivalent (g/amp-hr)
Enthalpy of Vaporization (kJ/mole)	Electron work function (V)
Brinell hardness (MN m ⁻²)	Pauling electronegativity

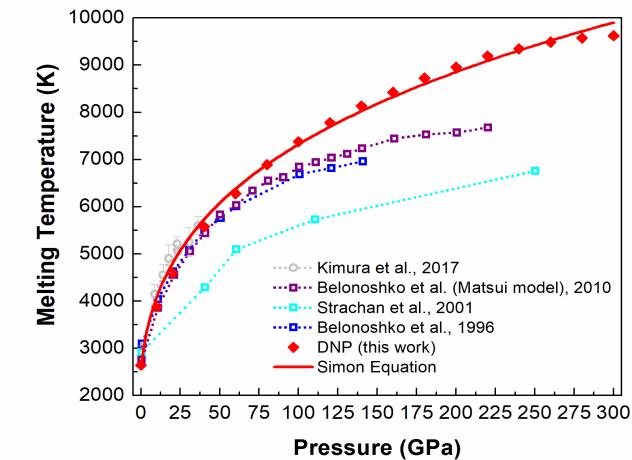
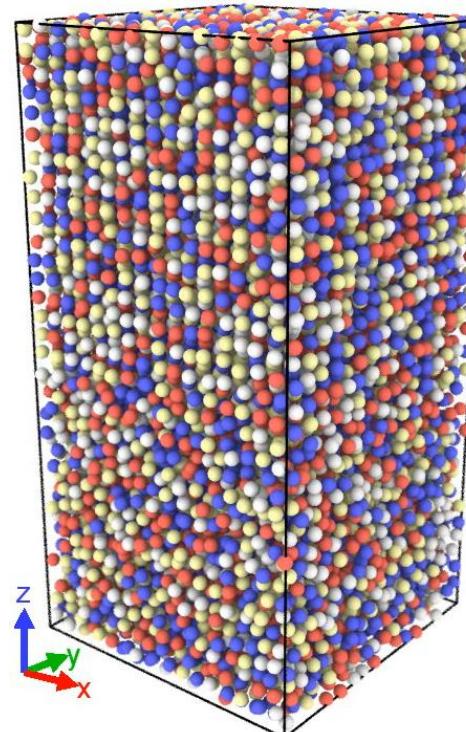
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Trehern, W., Ortiz-Ayala, R., Atli, K.C., Arroyave, R. and Karaman, I., 2022. Data-driven shape memory alloy discovery using artificial intelligence materials selection (AIMS) framework. *Acta Materialia*, 228, p.117751. <https://doi.org/10.1016/j.actamat.2022.117751>

Summary

- MLPs provide a good balance of accuracy and computational cost, and allows bridging the modeling gap with experiment
- Various examples including melting and early stages of oxidation of a binary alloy
- Creep rapture time prediction using experimental data and simple features



Database and MLPs are available
<https://github.com/saidigroup>

Disclaimer



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