

# Smart Refractory Sensors Development for Corrosion and Erosion Monitoring in High Temperature Systems

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**Corrosion Science and Technology**

# Introduction

- Processes such as energy generation, metals/glass manufacturing, coal gasification and aerospace technology applications require health and process monitoring in harsh-environments.
- Harsh-environments conditions include:**
  - ❖ High temperature (500-1800°C)
  - ❖ High pressure (up to 1000 psi)
  - ❖ Corrosive, erosive and reducing environments.
- Ability to monitor:**
  - ❖ Temperature
  - ❖ Structural stability of systems components.
- US DOE Overall Goal:** Develop health and temperature sensors (and sensor arrays) embedded into refractory compositions.



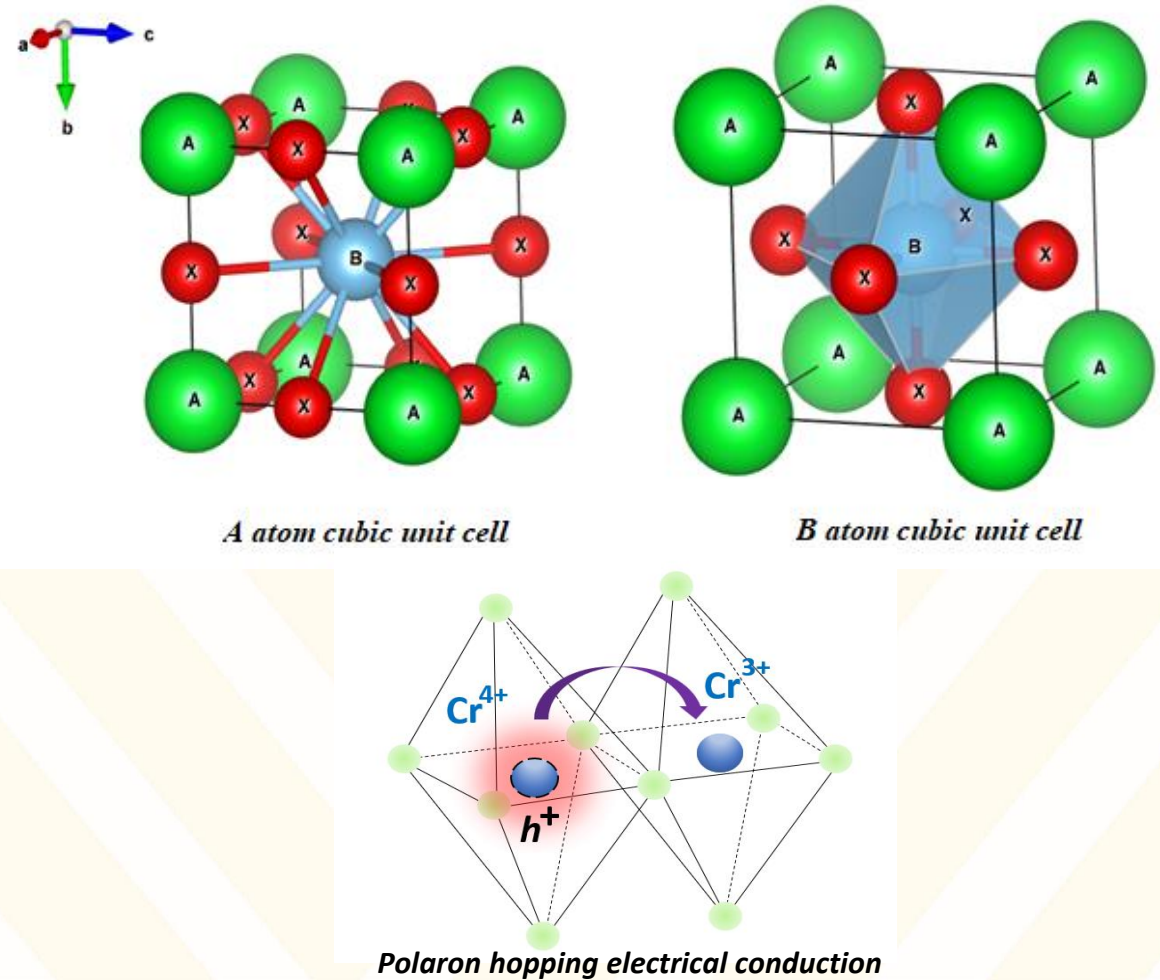
# ***Objectives of This Work***

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- ❖ Synthesize conductive refractory lanthanum oxides perovskites using the Pechini Sol-Gel method.
- ❖ Study the electrical conductivity and Seebeck coefficients of such compositions at high temperatures and under different working atmospheres (oxidizing, reducing).
- ❖ Fabricate surface printed thick-film sensors utilizing these materials and test at high-temperature.
- ❖ Fabricate sensors embedded into refractory and perform thermoelectrical and corrosion testing.

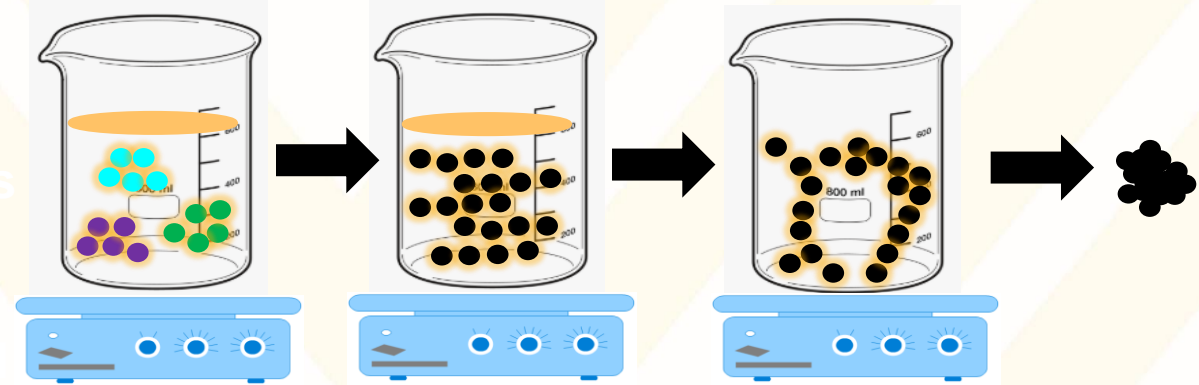
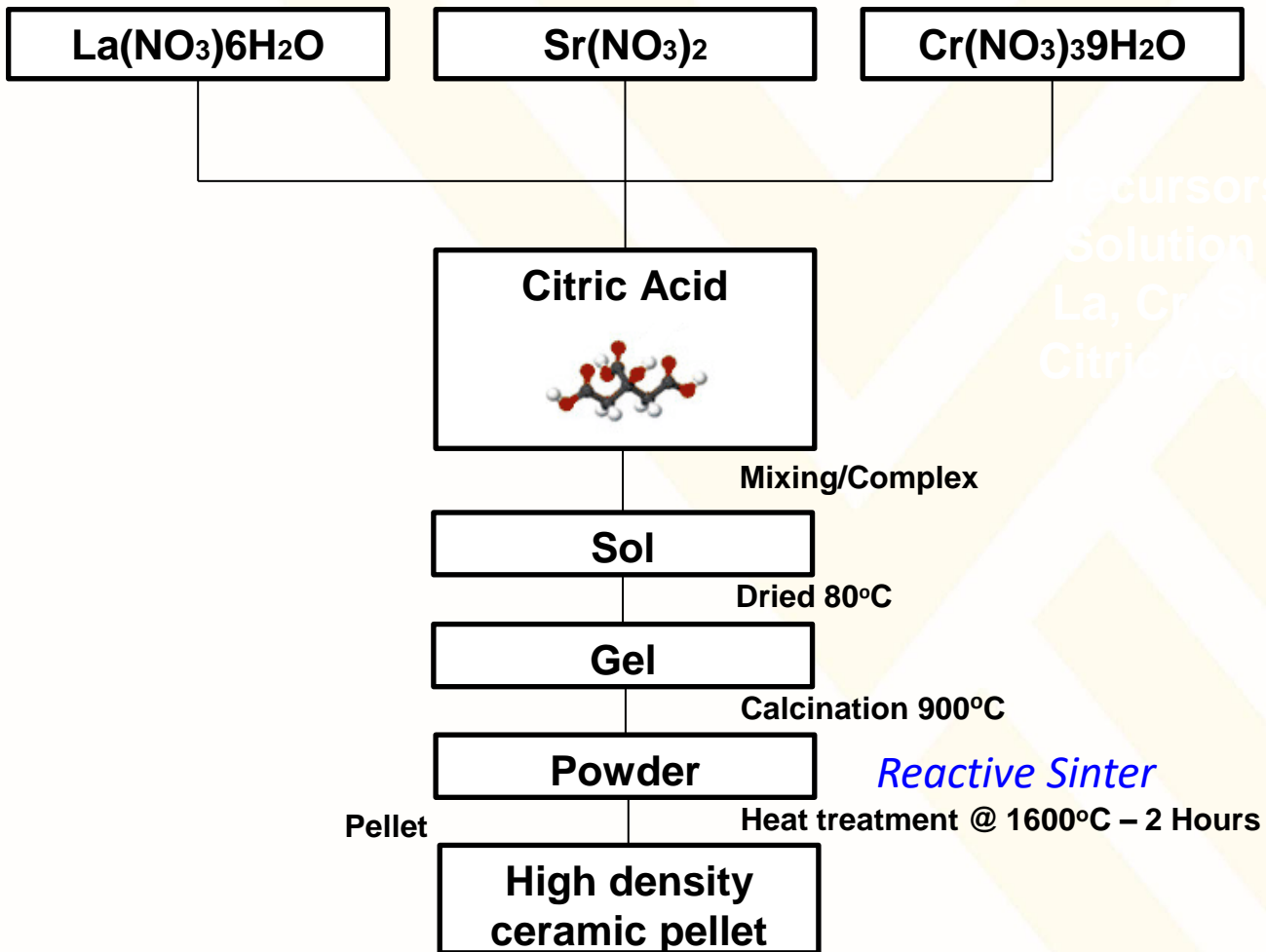
# Lanthanum Chromite: General Aspects

- ❖ High melting point ( $\sim 2500^\circ\text{C}$ ).
- ❖ Chemically stable under oxidative and reducing atmospheres.
- ❖ Pure  $\text{LaCrO}_3$  shows semiconducting behavior with no to low ionic conduction.
- ❖ Calcium substitution increase conductivity from 1.0 to  $40.0 \text{ S}\cdot\text{cm}^2$  at  $1000^\circ\text{C}$  (Mori *et al.* 1997)
- ❖ Compatibility (thermal expansion coefficients matching) near refractory materials,  $\sim 10 \times 10^{-6} \text{ }^\circ\text{C}^{-1}$ .





# Sol Gel Synthesis and Pellet Fabrication



- ✓ Pechini-like process used.
- ✓ High homogenous and adequate sintering.
- ✓ High density (typical in literature <93% density).
- ✗ Low yields and not easy to scale-up

## Compositions Studied:

**A-site:**  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$

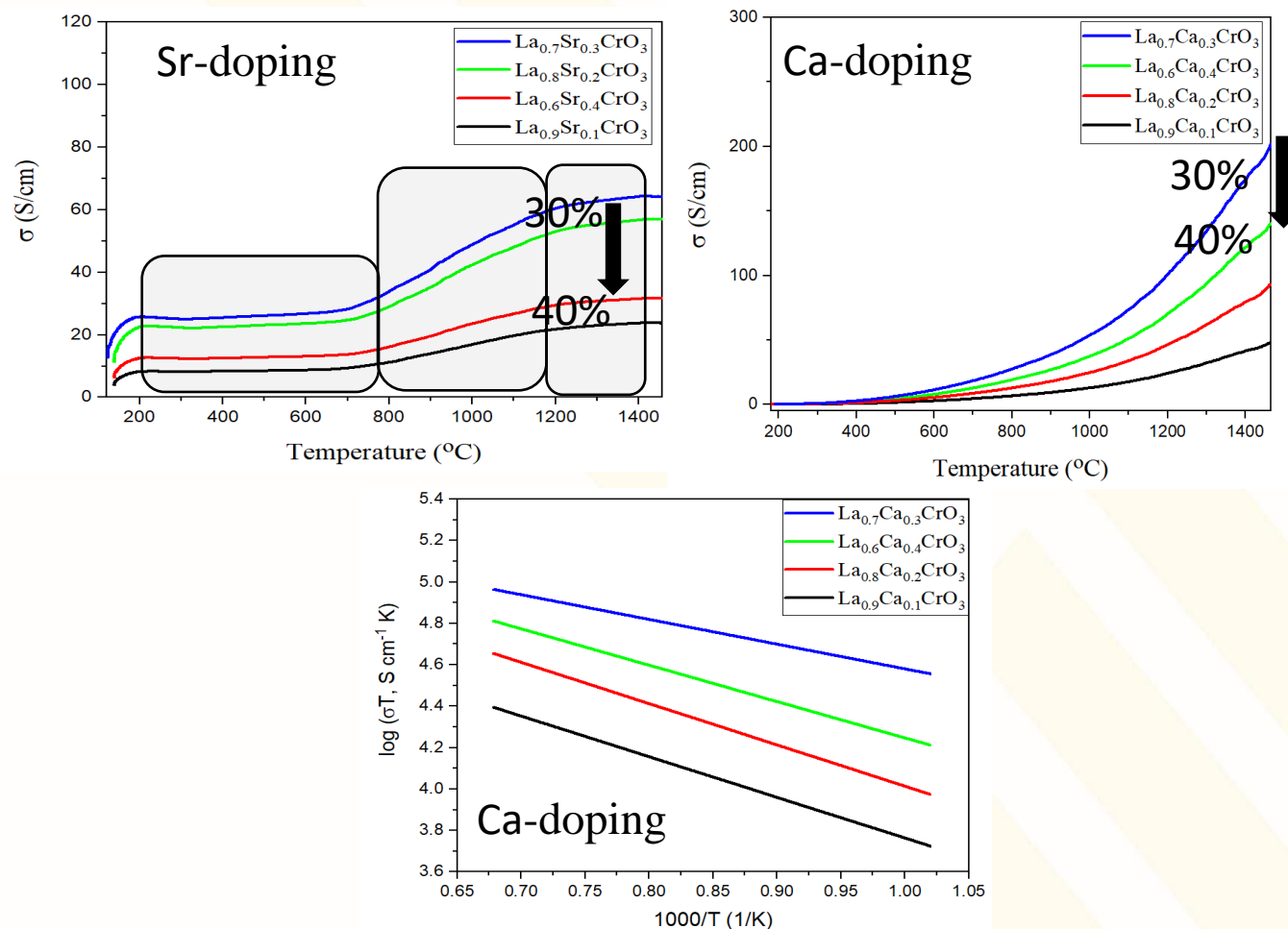
( $x = 0.1, 0.2, 0.3, 0.4$ )

**B-site:**  $\text{La}_{0.8}\text{Sr}_{0.2}\text{Cr}_{1-y}\text{Mn}_y\text{O}_3$

( $y = 0.1, 0.2, 0.3, 0.4$ )

# Electrical Conductivity Characterization

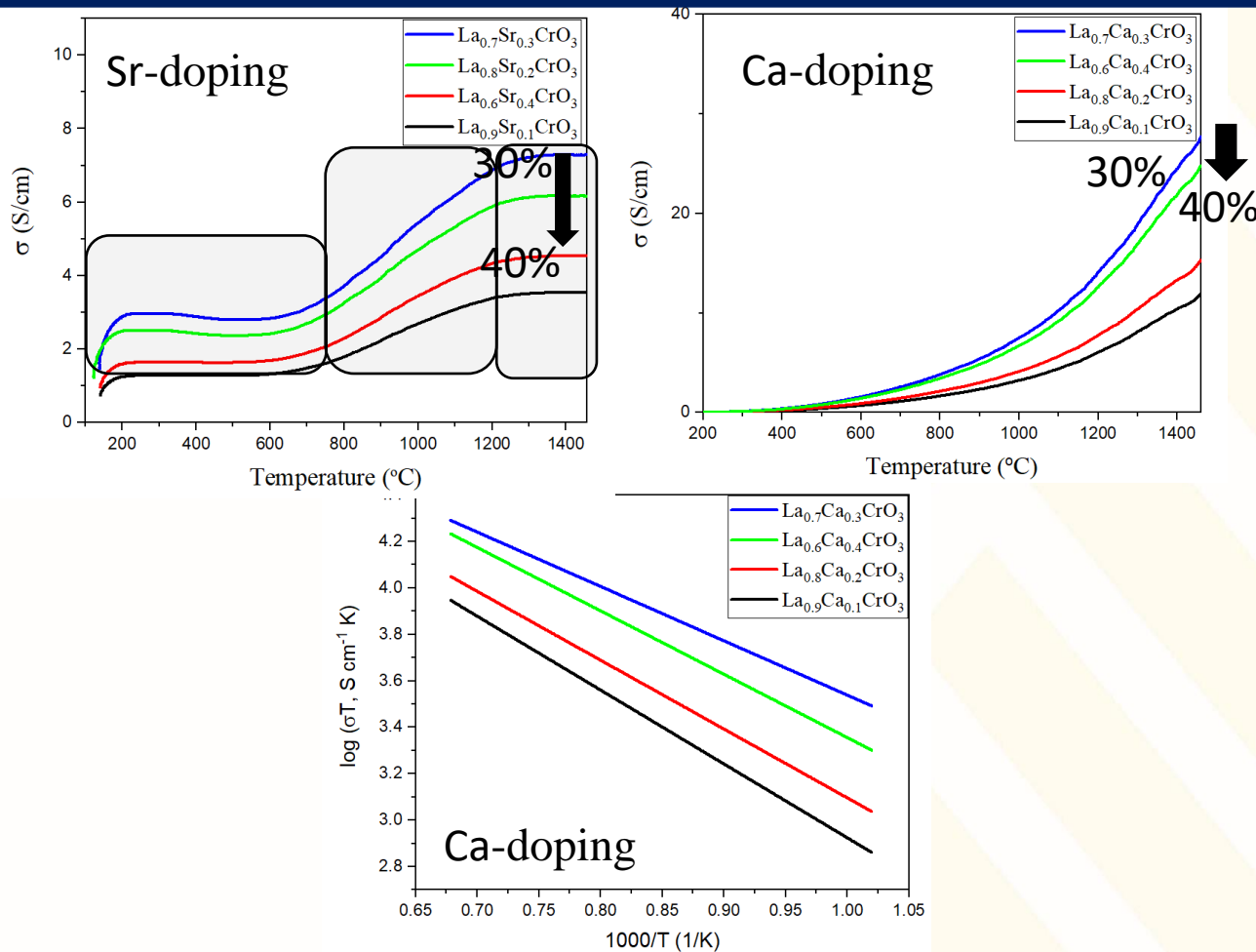
# DC Electrical Conductivity (Oxidizing Atmosphere)



- ❖ Conductivity typically exponentially increases with increase in carrier mobility, but 30 to 40% all drop in conductivity (believe slight second phase or higher lattice strain).
- ❖ Sr doped shows three regions, not seen in literature, since most tests  $<850^{\circ}\text{C}$ - $1000^{\circ}\text{C}$ . (believe  $V_o$  at high temperature)
- ❖ Arrhenius relationship fits for higher temperature regimes.
- ❖ Calcium doped compositions present higher conductivity due the lower distortion effects on lattice structure.

Electrical conductivity vs temperature and Arrhenius Plot  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$

# DC Electrical Conductivity (Reducing Atmosphere)



- ❖ Conductivity decrease for all temperature range under reducing atmosphere (H<sub>2</sub> 5% / N<sub>2</sub> 95%).
- ❖ Under reducing conditions, oxygen vacancies form to keep neutrality (drop in hole carrier concentration).
- ❖ Sr conductivity still displays regions of altered mechanism.
- ❖ All compositions present lower conductivity as dopant increases (but still 40% lower than 30% in all cases).

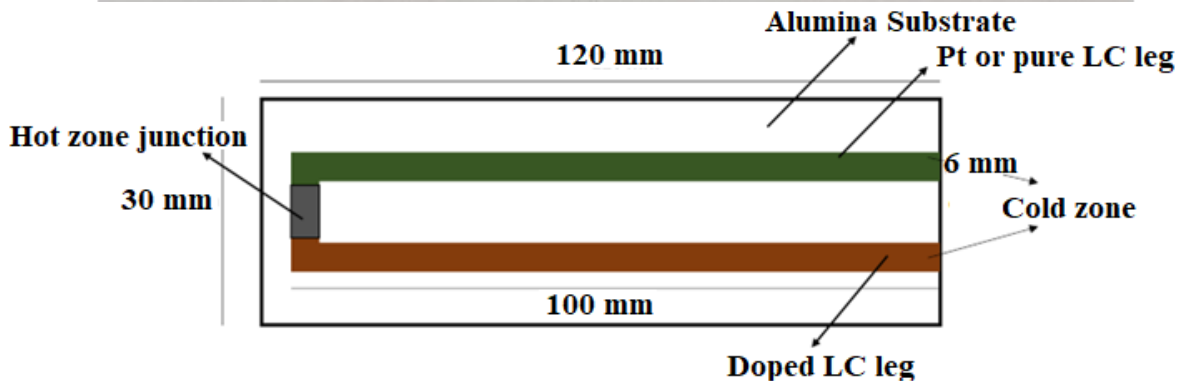
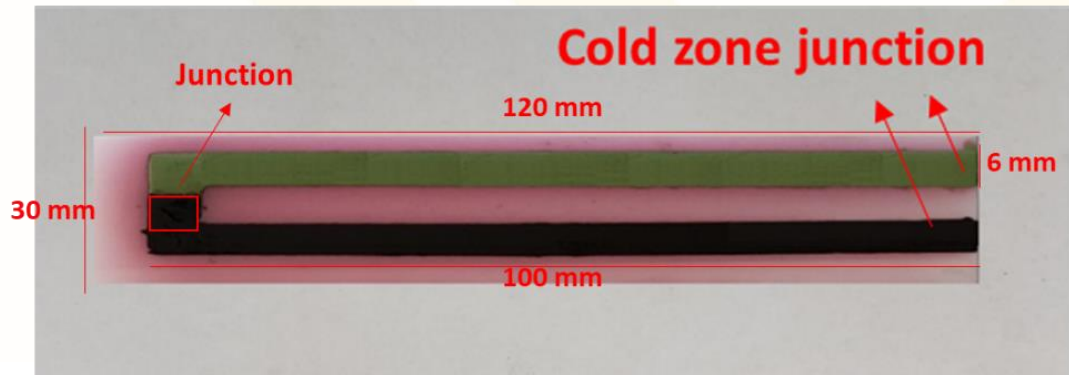
Electrical conductivity vs temperature and Arrhenius Plot for  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$



# Thick Film Sensors Fabrication

## Thermoelectrical Characterization

# High Temperature Sensors Fabrication



Leg 1 (Pt or pure-LaCrO<sub>3</sub>)

Leg 2 (Doped LaCrO<sub>3</sub>)

Ball milling in isopropanol for 8 h and drying

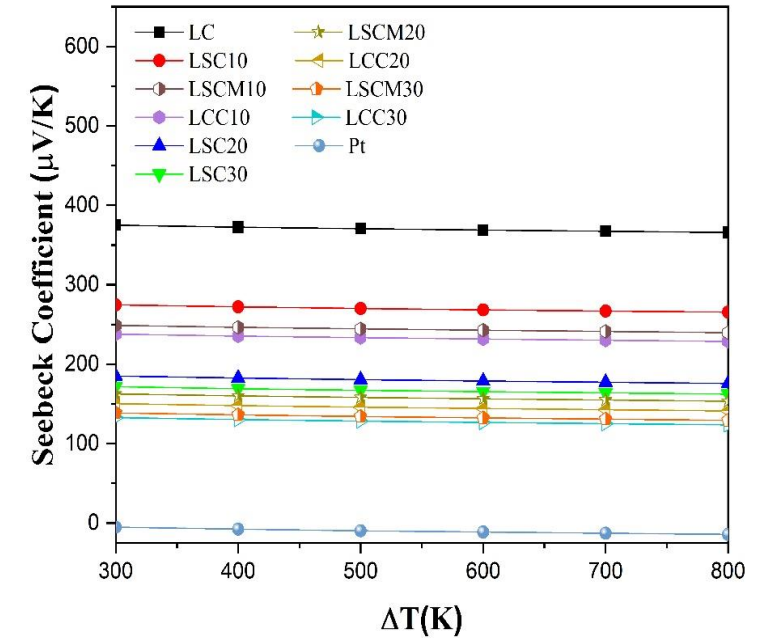
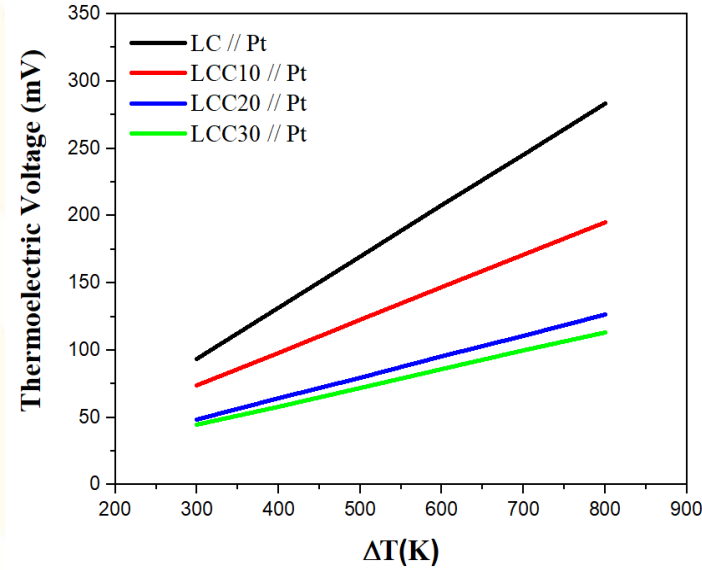
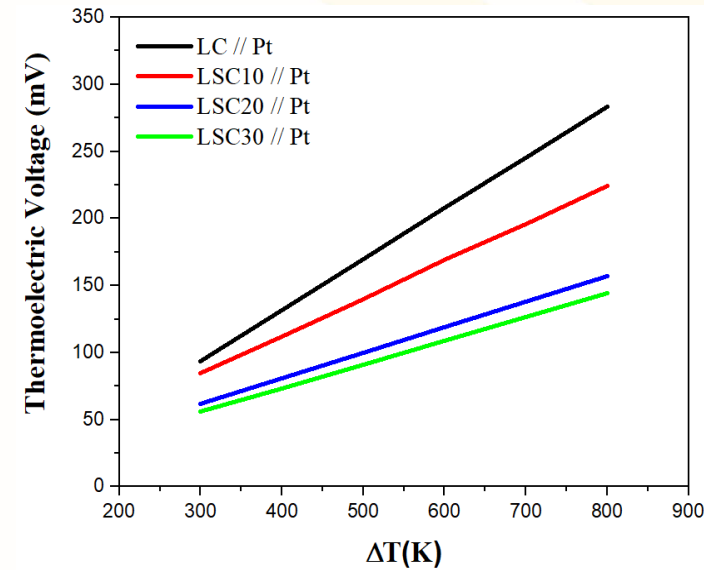
Ink preparation by mixing with an organic vehicle and ultrasonication

3D printing on as-prepared alumina substrates (120 x 30 mm) and drying

Sintering of the thermocouples (2°C/min, 1500°C, 1 h)

- ❖ Research team currently fabricating thermistors and thermocouples by 3D printing.
- ❖ High-temperature thermocouples that function >1200°C (in R-type range) new exciting development.

# Seebeck Coefficient Estimation (Using Pt Standard)

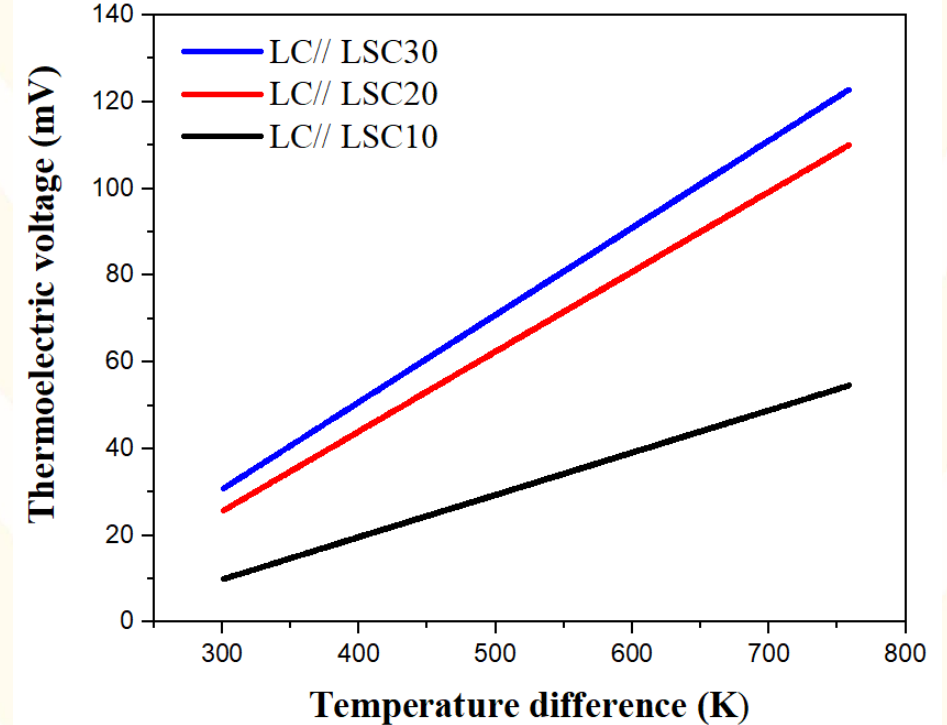
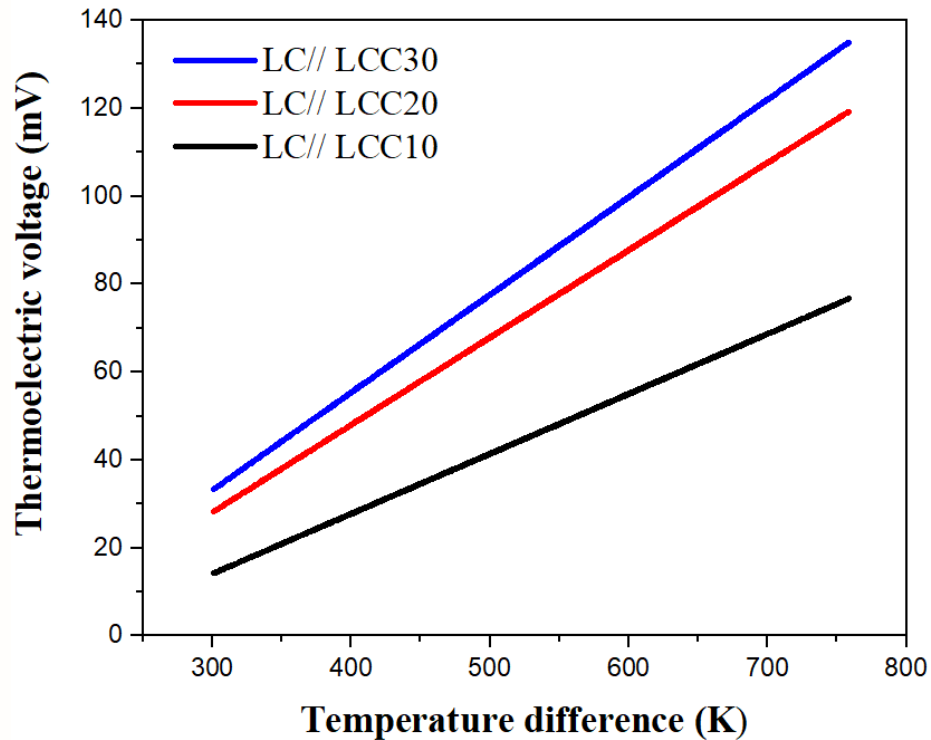


$$S(c) = (k_B/e) \ln[2(1 - c)/c]$$

**Heikes Equation**

- ❖ Linear correlation between temperature difference and thermoelectric voltage was observed for all the compositions.
- ❖ Doped-LaCrO<sub>3</sub>/Pt couples were fabricated to estimate absolute Seebeck coefficient ( $S_{Pt} \sim -18 \mu V/K^*$ ) up to 1000°C.
- ❖ Ca doping shows lowest absolute Seebeck coefficient with increasing Ca content.

# Thermoelectric Characterization of Thermocouples



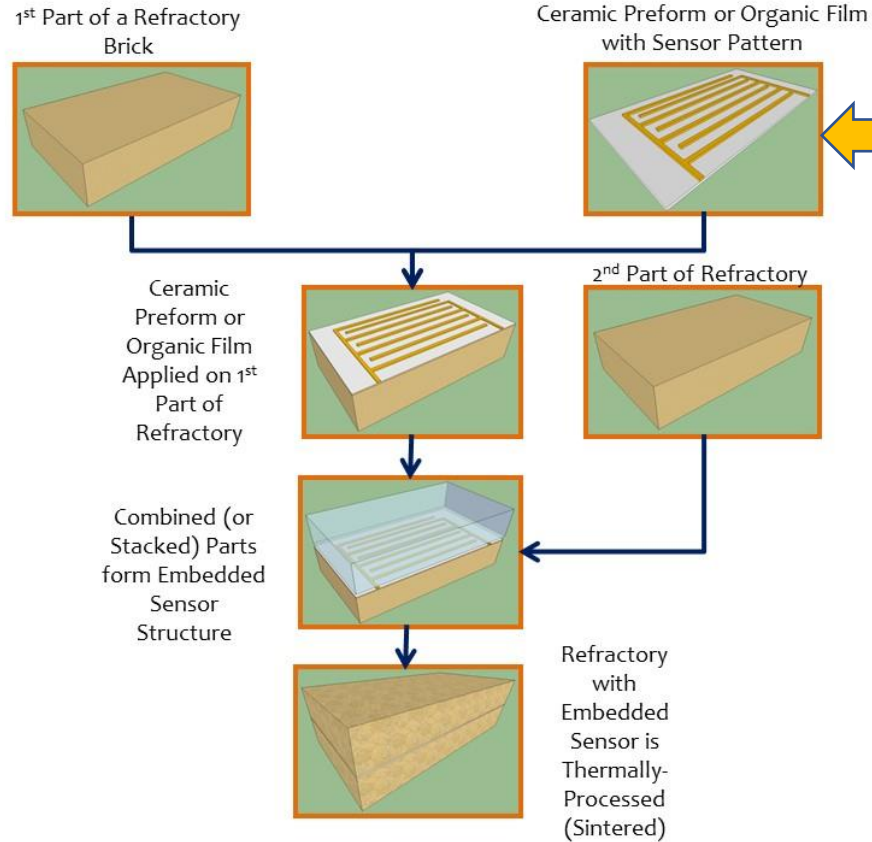
- ❖ Thermocouples were tested in a range between 30 to 850°C during 3 heating cycles, showing an excellent reproducibility.
- ❖ The LCC30/LC, LCC20/LC and LCC10/LC thermocouples showed a maximum higher voltage by **two orders of magnitude in comparison with Pt/Pt-Rh**, with values of 138.61 mV, 119.50 mV and 79.10 mV respectively (at  $\Delta T \sim 750^\circ\text{C}$ ).

# Sensors Embedded into Refractory

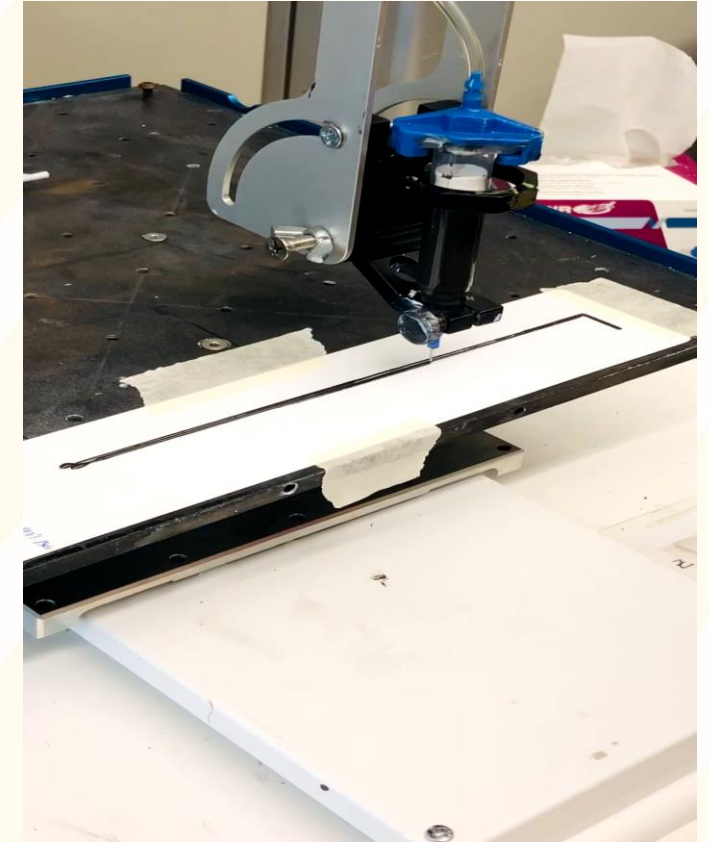
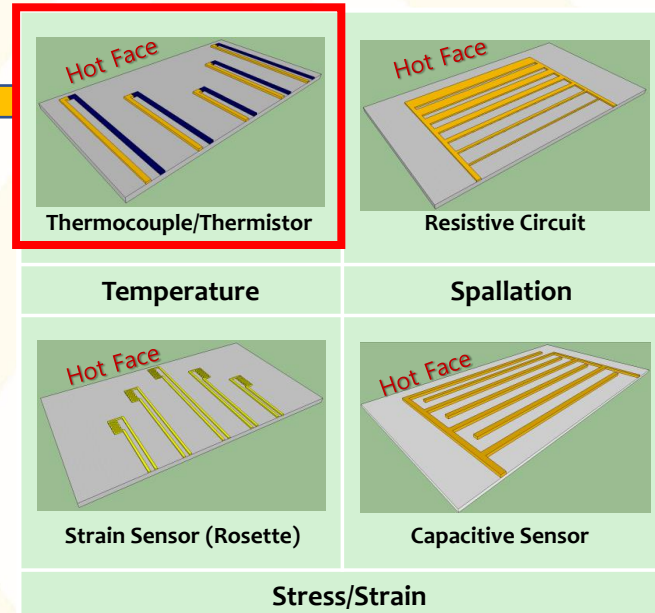


# Embedded Sensors Fabrication

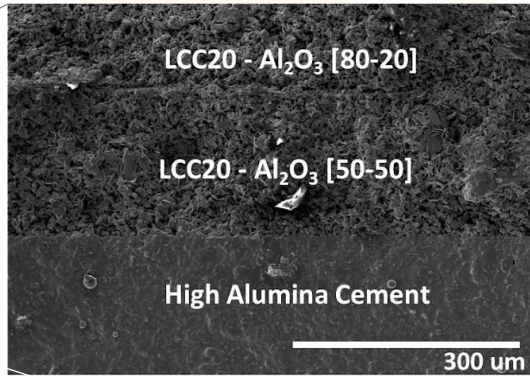
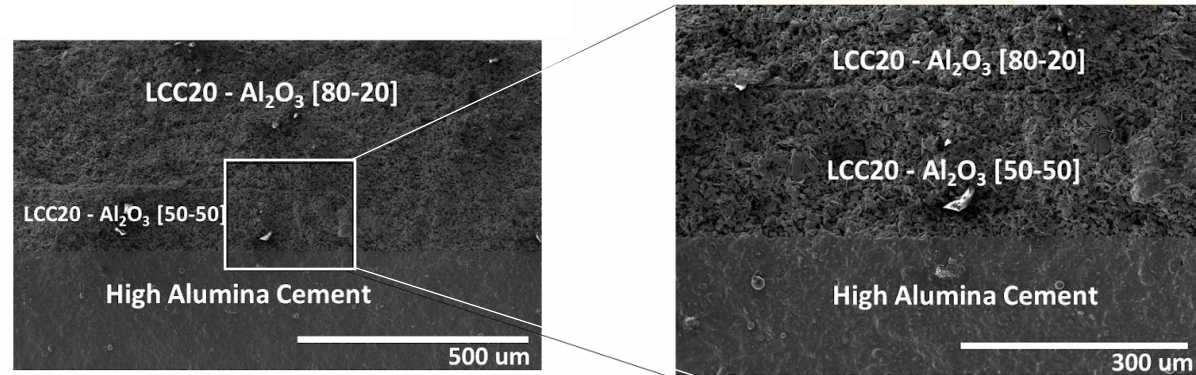
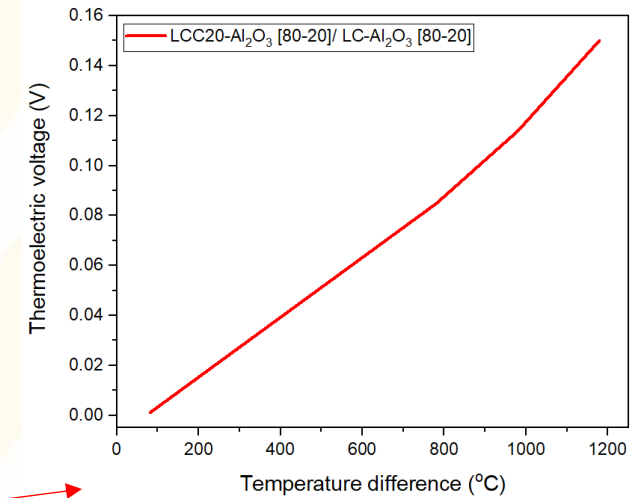
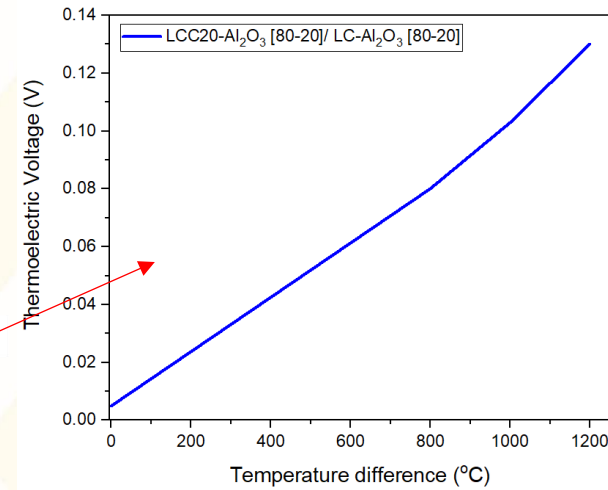
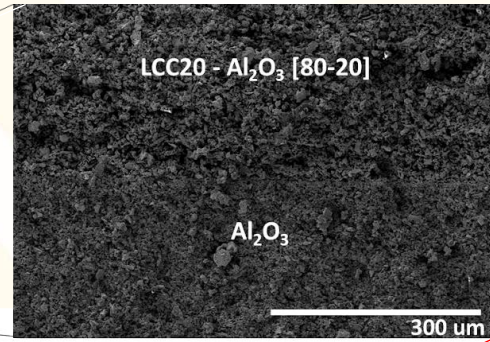
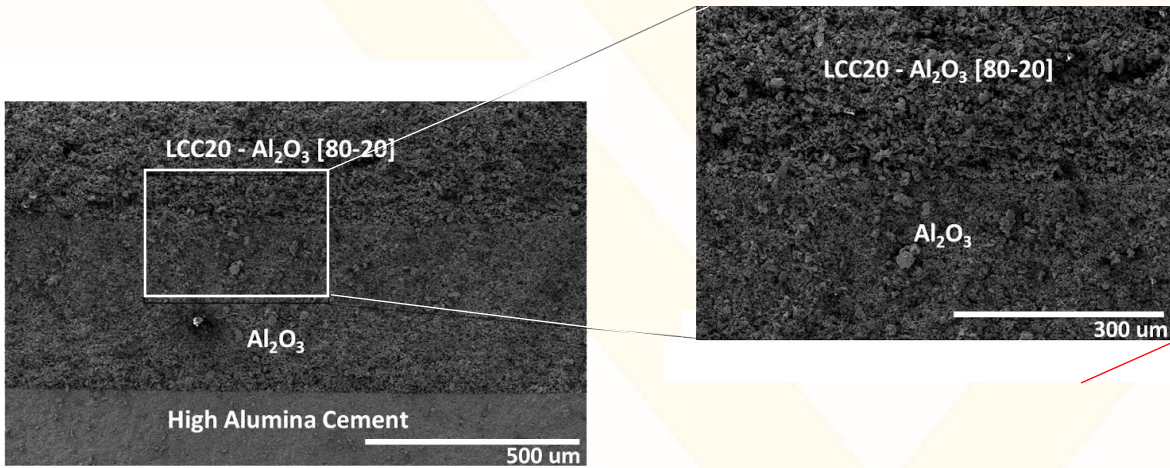
## General Smart Refractory Processing Method



## Examples of Sensor Preforms



# Embedded Thermocouples Characterization

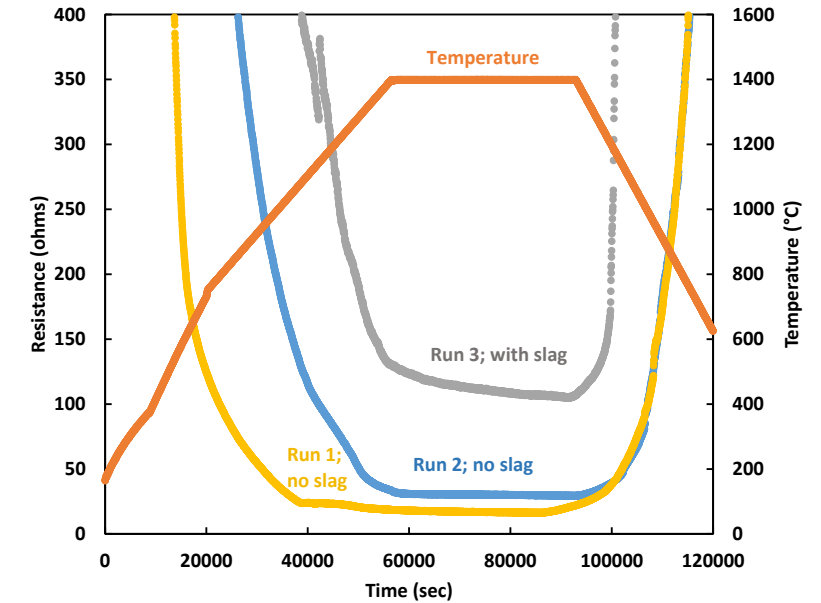
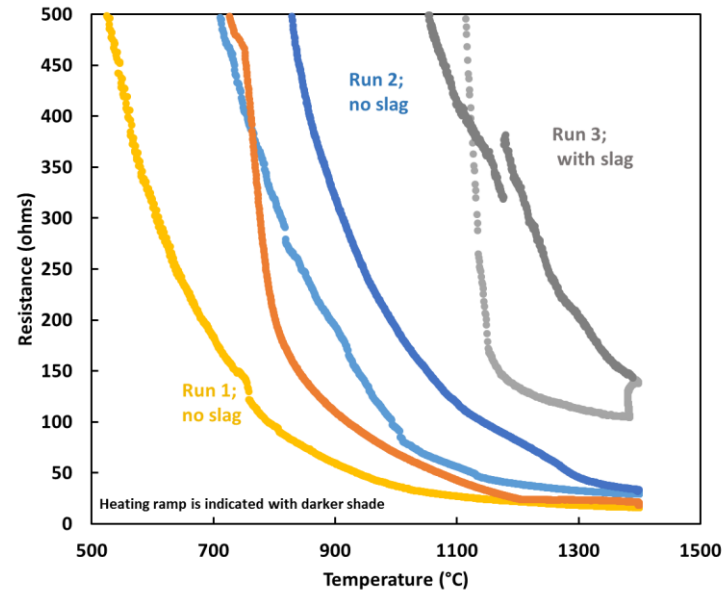
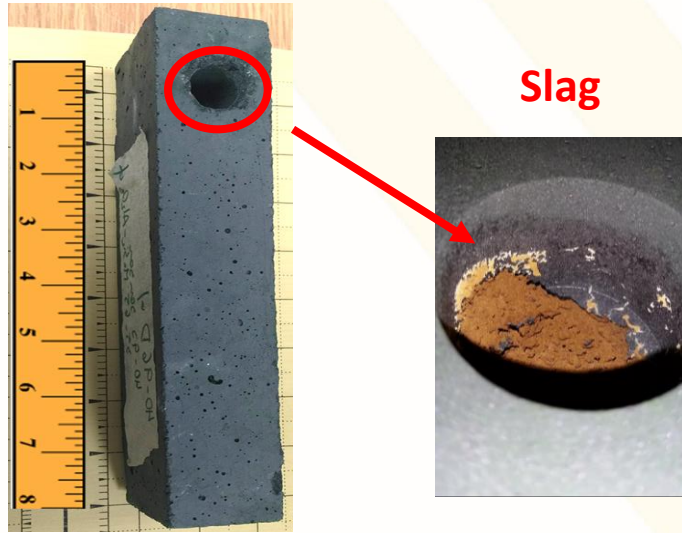


- ❖ Different 3D printing gradients were printed and embedded into high alumina refractory cement.
- ❖ Embedded thermocouples showed thermoelectric voltage responses  $\sim 0.15$  V in at  $\Delta T \sim 1200^{\circ}\text{C}$

# Embedded Sensors Initial Corrosion Testing



# Corrosion Testing of Embedded Sensors



Chromia refractory bricks

- ❖ Resistance and temperature profile and resistance versus time, respectively, for brick containing LNO (coarse) thermistor tested to 1400°C for 10 hrs.
- ❖ The brick was cycled between room temperature and 1400°C two times without slag. Before third run, 32 grams of slag pellets were added.
- ❖ With each thermal cycle, the resistance at each hold appears to increase, while it still shows slight decrease during the hold as shown with data collected with preforms.

# Conclusions



# Conclusions

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- ❖ **Electrical conductivity** shows correlated **dependence** with high **temperature (up to 1500°C)** for all compositions. The exponential Arrhenius trend is evidence for the **polaron hopping electrical conduction** mechanism.
- ❖ All the Seebeck coefficients were determined as the slope of the obtained plots, observing constant behavior as expected for **polaron hopping** active semiconductors such as doped  $\text{LaCrO}_3$ .
- ❖ It was observed that Seebeck coefficient **reduces** with the **increase** in **dopant** substituents as expected by **Heikes model**.
- ❖ **Chromium diffusion** was observed, there is a necessity for **protect conductive layers** and reduce chromium lost.

# Conclusions

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- ❖ **Embedded thermocouples** into oxide refractories were fabricated using an easy approach.
- ❖ The **embedded thermocouples** showed a correlated **increase in thermoelectric voltage** (as expected) in function of **temperature difference increment**.
- ❖ The **change in resistance** of the embedded thermocouples in time can be related to **platinum connections** at high temperatures and multiple testing.

# Acknowledgment

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- ❖ We would like to thank **U.S. Department of Energy (DOE)** and Maria Reidpath (project manager) for sanctioning this project **DE-FE0031825**.
- ❖ We also would like to acknowledge Mr. Harley Hart, Dr. Qiang Wang, and Dr. Marcela Redigolo for their cooperation and valuable assistance in the WVU Shared Research Facilities (SRF).
- ❖ We also would like to thank HWI, for support us in developing real-life applications sensing systems/devices.
- ❖ Kindly acknowledge faculty and staff of West Virginia University for their support.



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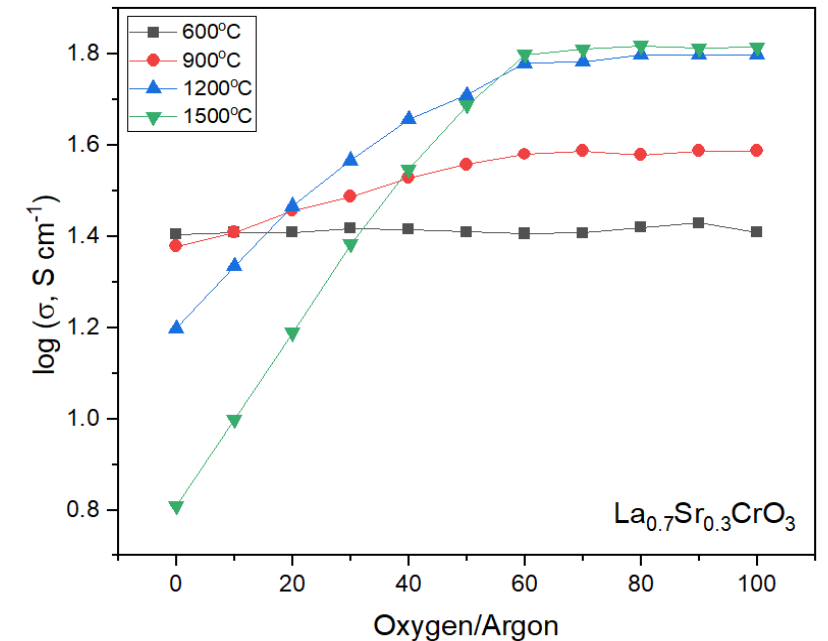
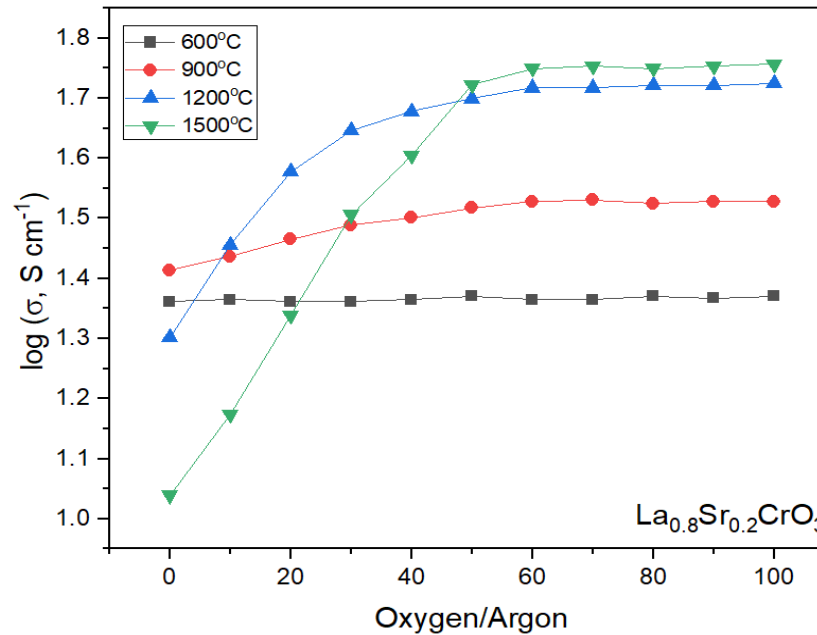
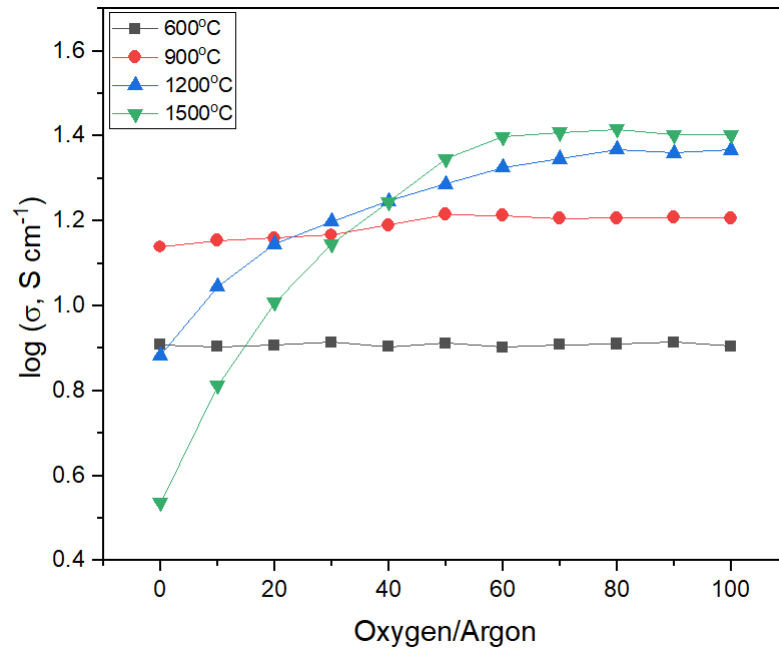
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# Electrical Conductivity (Oxygen partial pressure)

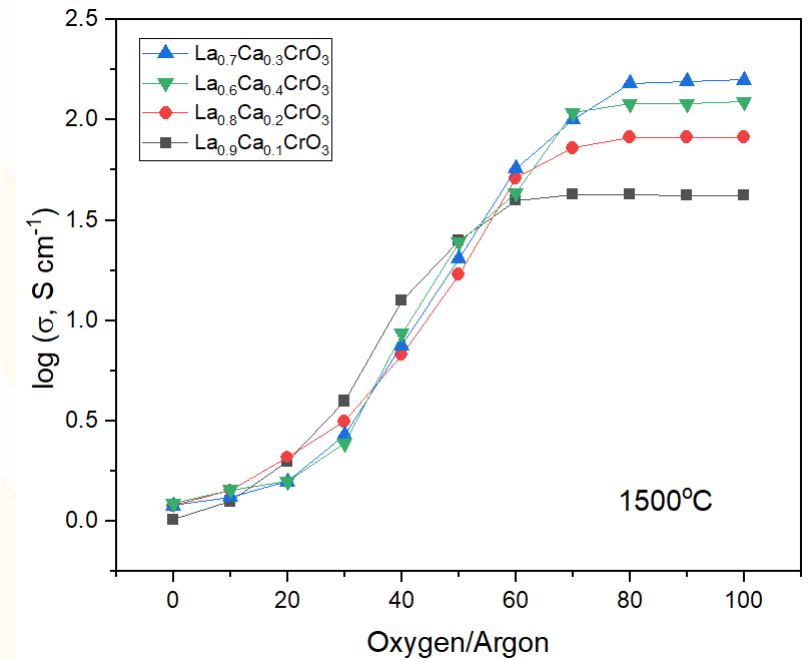
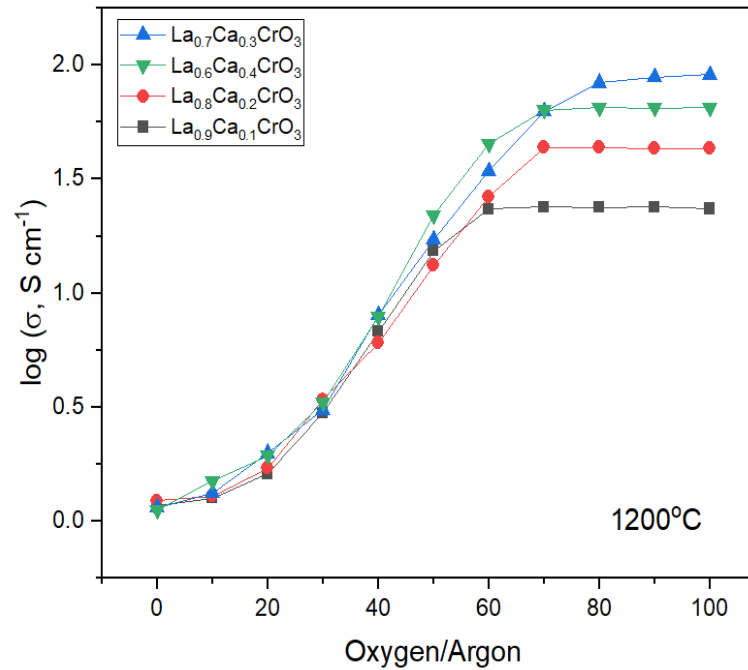
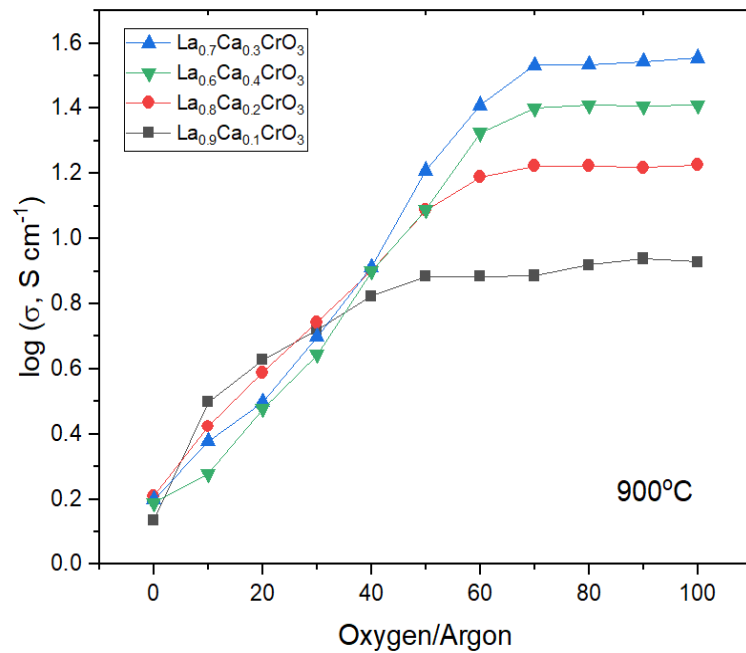


Electrical conductivity dependence of oxygen partial pressures at different temperatures for  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$

- ❖ At lower temperatures (600°C - 900°C) not significant changes in conductivity occurs for lanthanum doped compositions during the equilibrium time used (90 minutes).
- ❖ At higher temperatures (1200°C and 1500°C) the conductivity drops exponentially at lower oxygen partial pressures. Increasing the strontium concentration, the conductivity drop significantly.



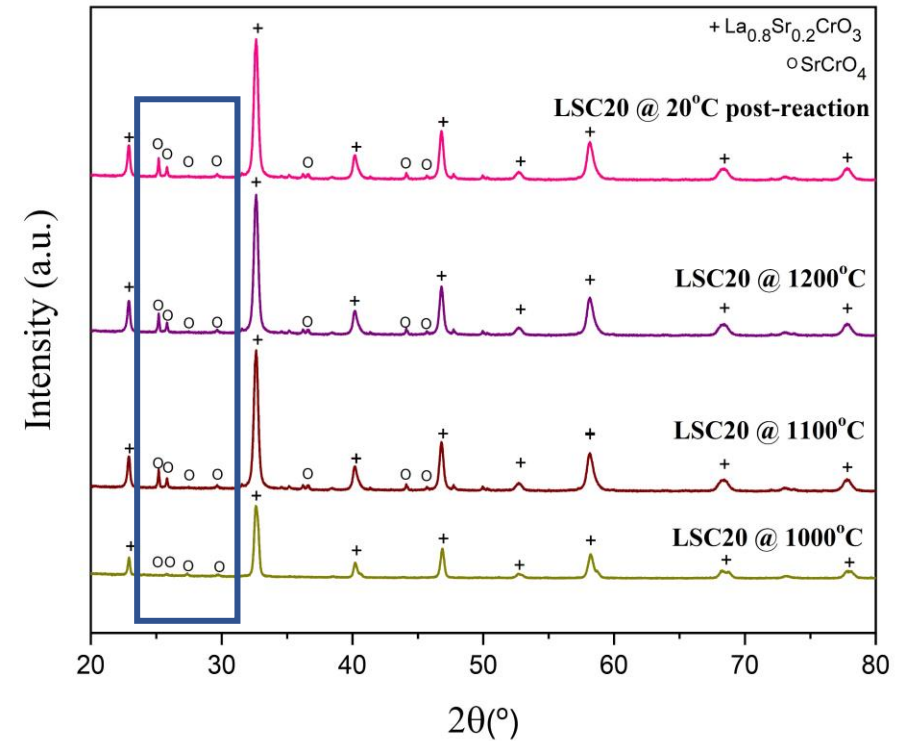
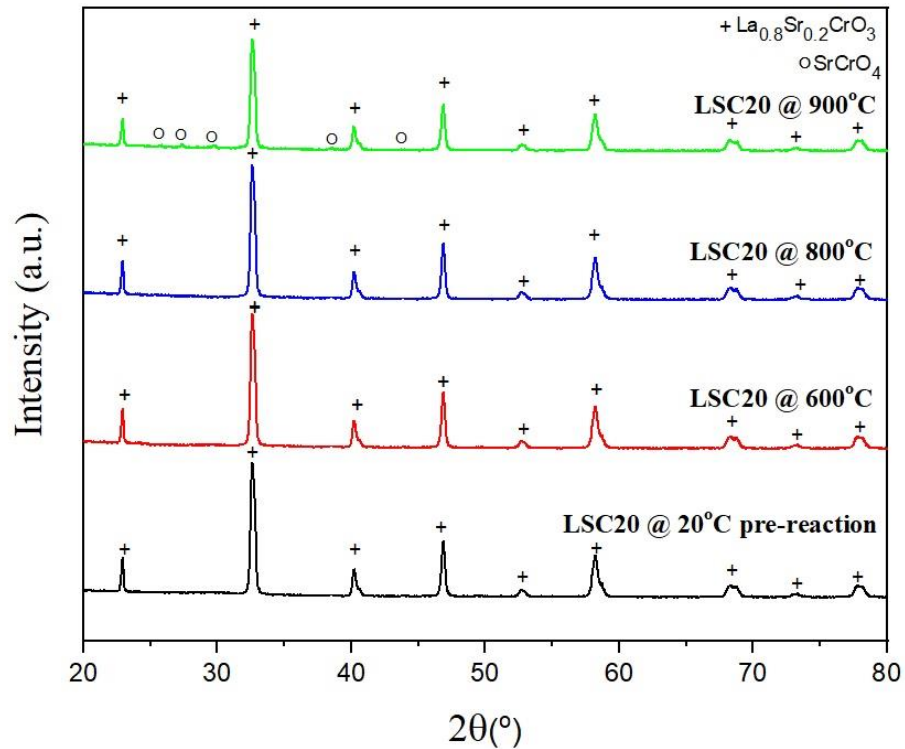
# Electrical Conductivity (Oxygen partial pressure)



Electrical conductivity dependence of oxygen partial pressures at different temperatures for  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$

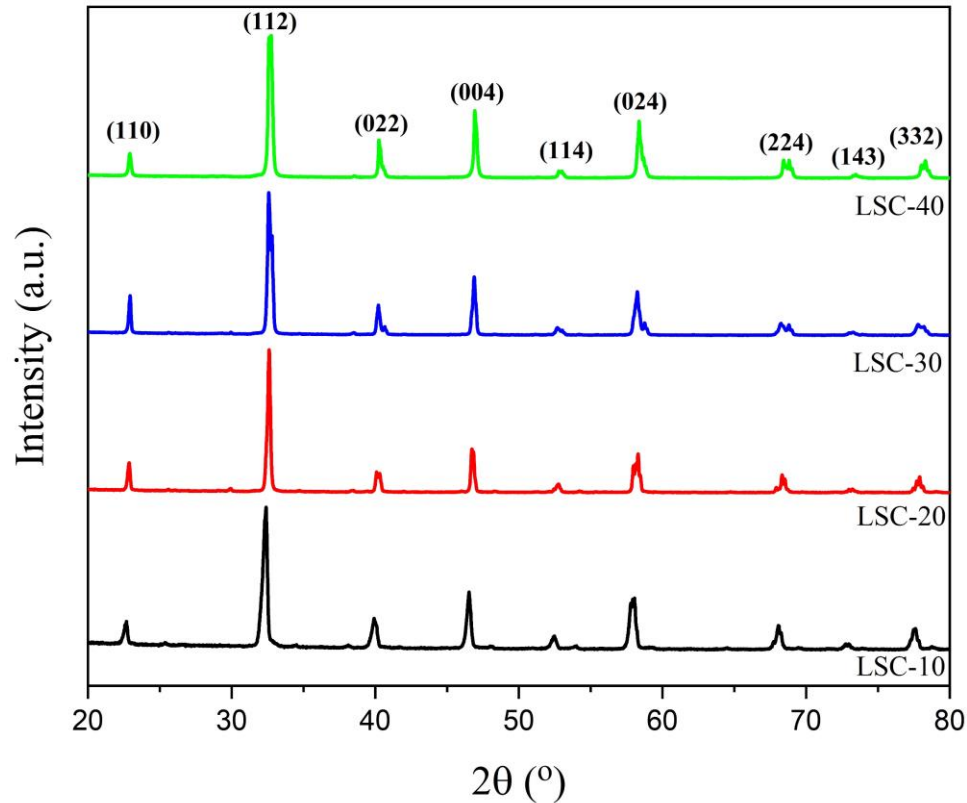
- ❖ When the oxygen partial pressure goes below a critical value, the oxygen vacancies are generated at expense of electron holes and conductivity decrease for all compositions.
- ❖ The charge imbalance caused by the introduction of Calcium starts to be compensated by the formation of oxygen vacancies.

# ***Sr doped lanthanum chromite stability experiments***



***X-ray diffractograms of 20% strontium doped lanthanum chromite annealed at different temperatures.***

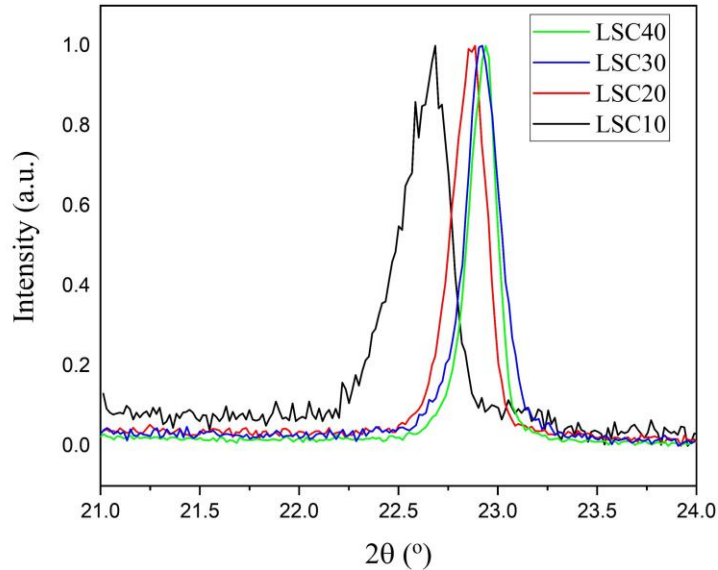
# Crystalline Structure/Phase Analysis



X-ray diffractograms for the samples of the  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$  series

- ❖ Single phase doped lanthanum chromites materials were obtained successful (no residual oxide or pyrochlore peaks).
- ❖ Using Pechini Sol-Gel method permitted doped lanthanum chromites at high solubility levels (40%).
- ❖ Solubility limits >40% substitution level (Sujatha *et al.* 1992).
- ❖ No impurities extra peaks were present in the final prepared powders and ceramic pellets.

# Lattice Parameters, Unit Cell Volume and Density



X-ray diffractograms showing shifting of (110) peak for the  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$  series

Lattice parameters, unit cell volume and XRD theoretical density for doped lanthanum chromites perovskites

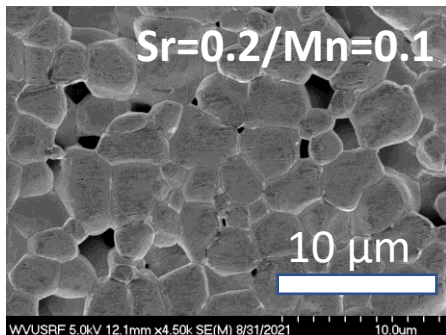
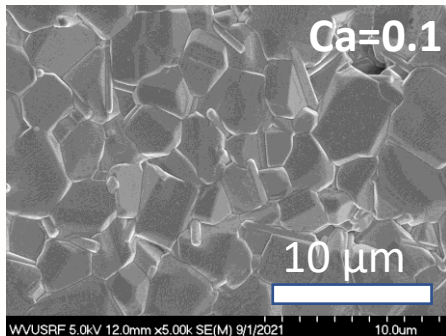
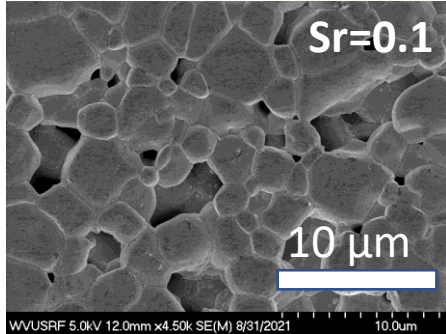
Composition	Lattice parameters (Å)			Volume (Å <sup>3</sup> )	$\rho_{\text{XRD Theoretical}}$ (g/cm <sup>3</sup> )
	a	b	c		
$\text{La}_{0.9}\text{Sr}_{0.1}\text{CrO}_3$	5.5124	5.5668	7.7926	239.1299	6.4932
$\text{La}_{0.8}\text{Sr}_{0.2}\text{CrO}_3$	5.4988	5.5425	7.7853	237.2747	6.4004
$\text{La}_{0.7}\text{Sr}_{0.3}\text{CrO}_3$	5.4769	5.5233	7.7580	234.6839	6.3259
$\text{La}_{0.6}\text{Sr}_{0.4}\text{CrO}_3$	5.4524	5.5122	7.7407	232.6441	6.2350
$\text{La}_{0.9}\text{Ca}_{0.1}\text{CrO}_3$	5.4180	5.5039	7.7332	230.6050	6.5963
$\text{La}_{0.8}\text{Ca}_{0.2}\text{CrO}_3$	5.4092	5.4982	7.7264	229.7898	6.3341
$\text{La}_{0.7}\text{Ca}_{0.3}\text{CrO}_3$	5.3994	5.4877	7.7058	228.3264	6.0872
$\text{La}_{0.6}\text{Ca}_{0.4}\text{CrO}_3$	5.3897	5.4622	7.6853	226.2520	5.8528
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.90}\text{Mn}_{0.10}\text{O}_3$	5.4734	5.5648	7.7765	236.8595	6.1533
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.80}\text{Mn}_{0.20}\text{O}_3$	5.4705	5.5587	7.7702	236.2829	6.1766
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.70}\text{Mn}_{0.30}\text{O}_3$	5.4598	5.5398	7.7498	234.4020	6.2344
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.60}\text{Mn}_{0.40}\text{O}_3$	5.4146	5.4981	7.7065	229.4225	6.3783

%Sr↑  
↓  
%Ca↑  
↓  
%Mn↑

- ❖ Decrease in lattice parameters (and volume) were observed when dopant cations is introduced in the lattice.
- ❖ To achieve neutrality chromium change from  $\text{Cr}^{+3}$  to  $\text{Cr}^{+4}$ , reduction in the chromium size occurs (Hyun Choi *et al.* 2013) .



# Microstructure/Grain Size Distribution



Average grain size and bulk density distribution for  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$  series

Composition	Average Grain Size ( $\mu\text{m}$ )	Relative Percentage Bulk Density (%)	
$\text{La}_{0.9}\text{Sr}_{0.1}\text{CrO}_3$	3.6	94	%Sr↑
$\text{La}_{0.8}\text{Sr}_{0.2}\text{CrO}_3$	3.5	95	
$\text{La}_{0.7}\text{Sr}_{0.3}\text{CrO}_3$	3.6	95	
$\text{La}_{0.6}\text{Sr}_{0.4}\text{CrO}_3$	3.2	94	
$\text{La}_{0.9}\text{Ca}_{0.1}\text{CrO}_3$	4.1	96	%Ca↑
$\text{La}_{0.8}\text{Ca}_{0.2}\text{CrO}_3$	3.7	97	
$\text{La}_{0.7}\text{Ca}_{0.3}\text{CrO}_3$	3.7	97	
$\text{La}_{0.6}\text{Ca}_{0.4}\text{CrO}_3$	3.6	98	

- ❖ Pechini Sol Gel prepared calcium, strontium, manganese doped lanthanum chromite powders exhibit better sinterability and densification under oxidizing conditions (undoped ↓90%).
- ❖ The samples of Ca doped lanthanum chromite powder have more dense microstructures. Furthermore, it was found that the incorporation of Ca, Sr in the A site of the lanthanum chromite increases grain growth (undoped <3  $\mu\text{m}$ ).



# DC Electrical Conductivities/Activation Energies

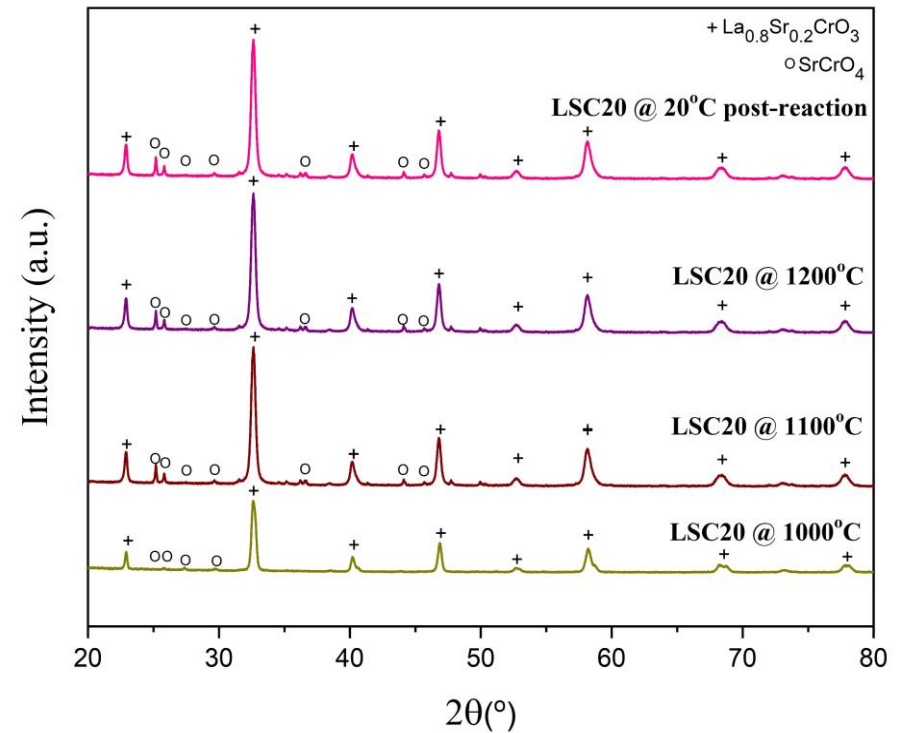
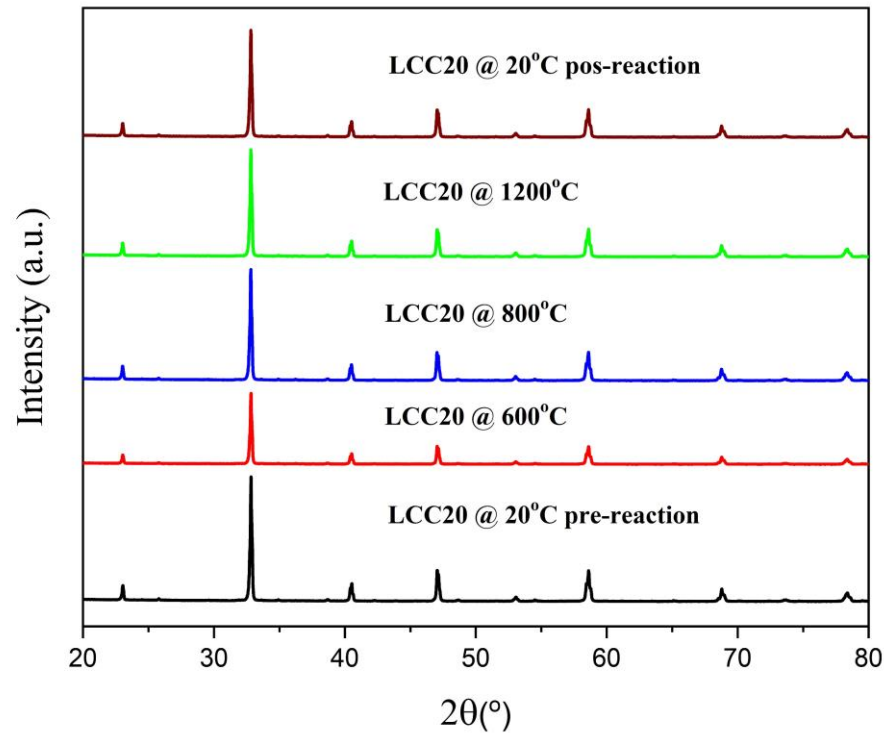
Composition	Air Atmosphere		Reducing Atmosphere	
	Conductivity @ 1000°C (S/cm)	Activation energy (eV)	Conductivity @ 1000°C (S/cm)	Activation energy (eV)
La <sub>0.9</sub> Sr <sub>0.1</sub> CrO <sub>3</sub>	16.672	0.1552	2.691	0.3238
La <sub>0.8</sub> Sr <sub>0.2</sub> CrO <sub>3</sub>	42.882	0.1427	4.572	0.2597
La <sub>0.7</sub> Sr <sub>0.3</sub> CrO <sub>3</sub>	49.032	0.1055	5.534	0.1719
La <sub>0.6</sub> Sr <sub>0.4</sub> CrO <sub>3</sub>	23.599	0.1498	3.462	0.3102
La <sub>0.9</sub> Ca <sub>0.1</sub> CrO <sub>3</sub>	13.211	0.1417	3.152	0.3240
La <sub>0.8</sub> Ca <sub>0.2</sub> CrO <sub>3</sub>	24.170	0.1298	4.118	0.2103
La <sub>0.7</sub> Ca <sub>0.3</sub> CrO <sub>3</sub>	52.823	0.1028	7.602	0.1367
La <sub>0.6</sub> Ca <sub>0.4</sub> CrO <sub>3</sub>	38.152	0.1175	6.626	0.1747

$$\sigma = \frac{\sigma_0}{T} \exp\left(\frac{-\Delta E_a}{kT}\right)$$

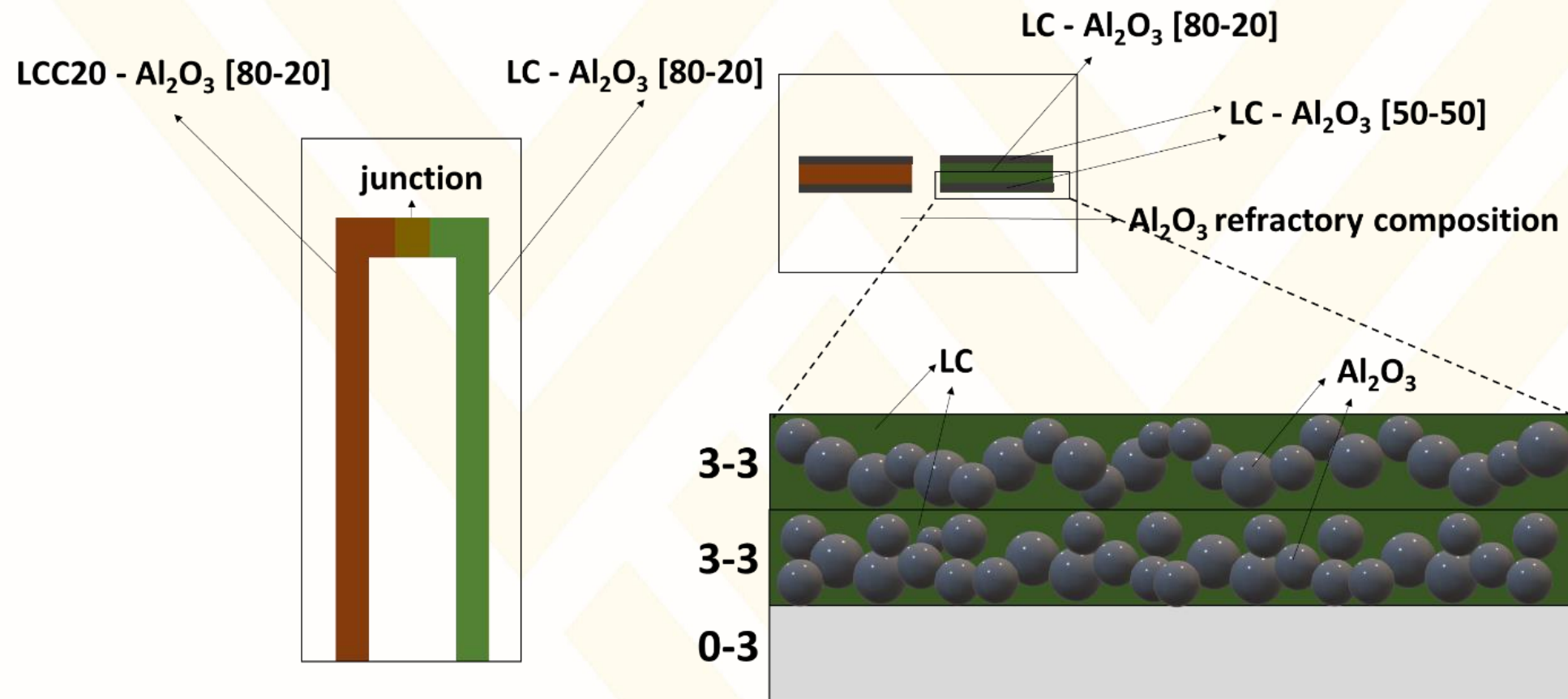
Slope  $\propto E_a$

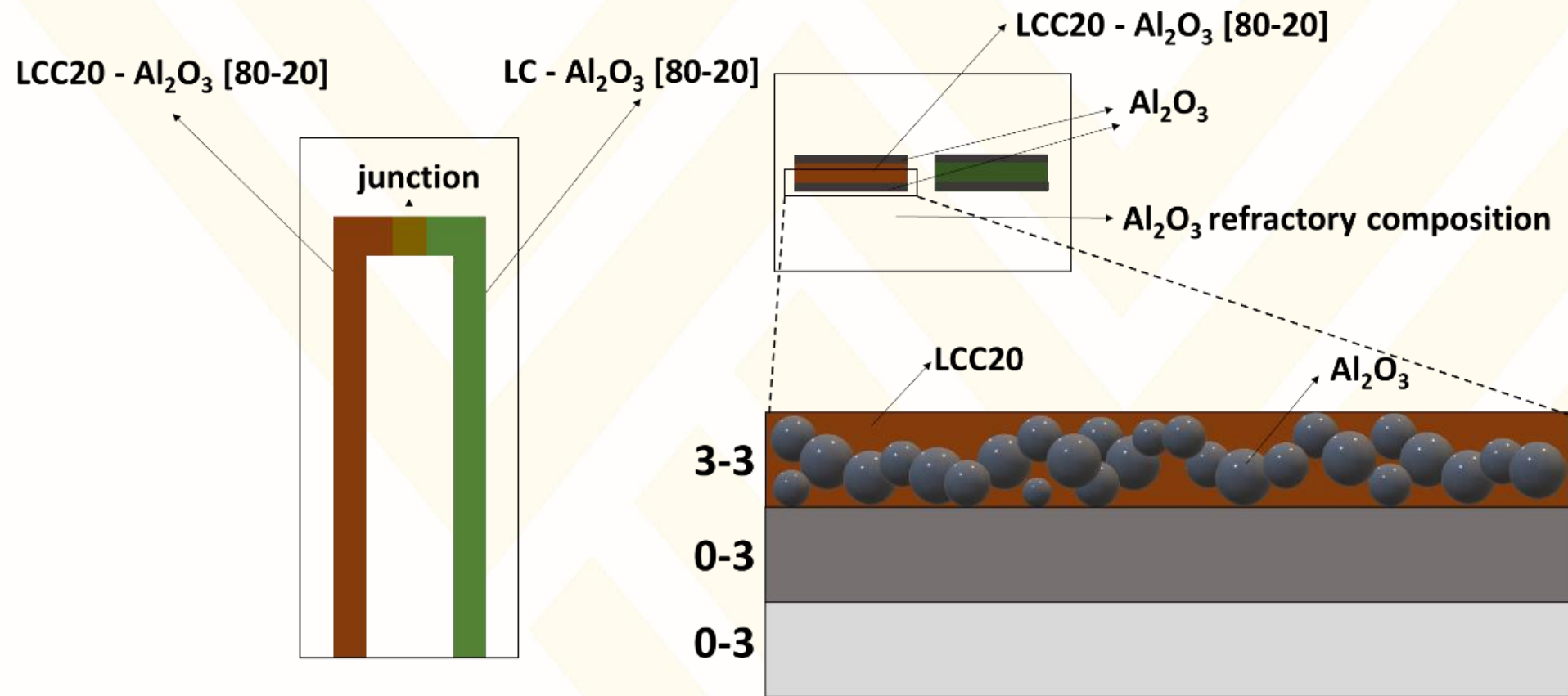
- ❖ Conductivity increase as function of doping level up to 30% for all dopants (strontium, calcium and manganese).
- ❖ At 40% doping levels conductivity decrease due to higher lattice distortion in all systems (solubility limit).
- ❖ Lower conductivity values under reducing atmospheres are explained by oxygen vacancies formation (near  $>1.5\times$  in activation energy).

# ***Ca doped lanthanum chromite stability experiments***

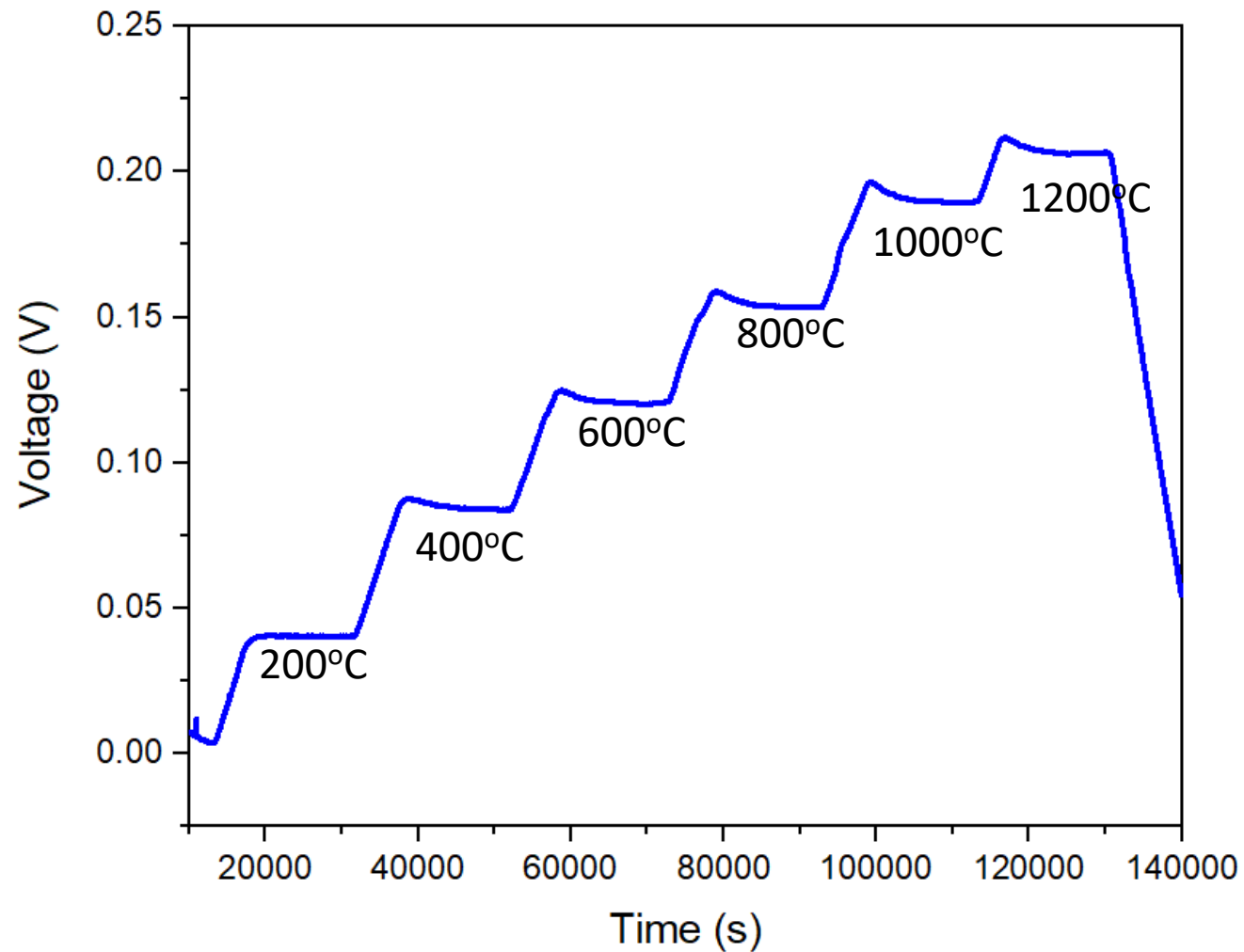


***X-ray diffractograms of 20% strontium and calcium doped lanthanum chromite annealed at different temperatures.***



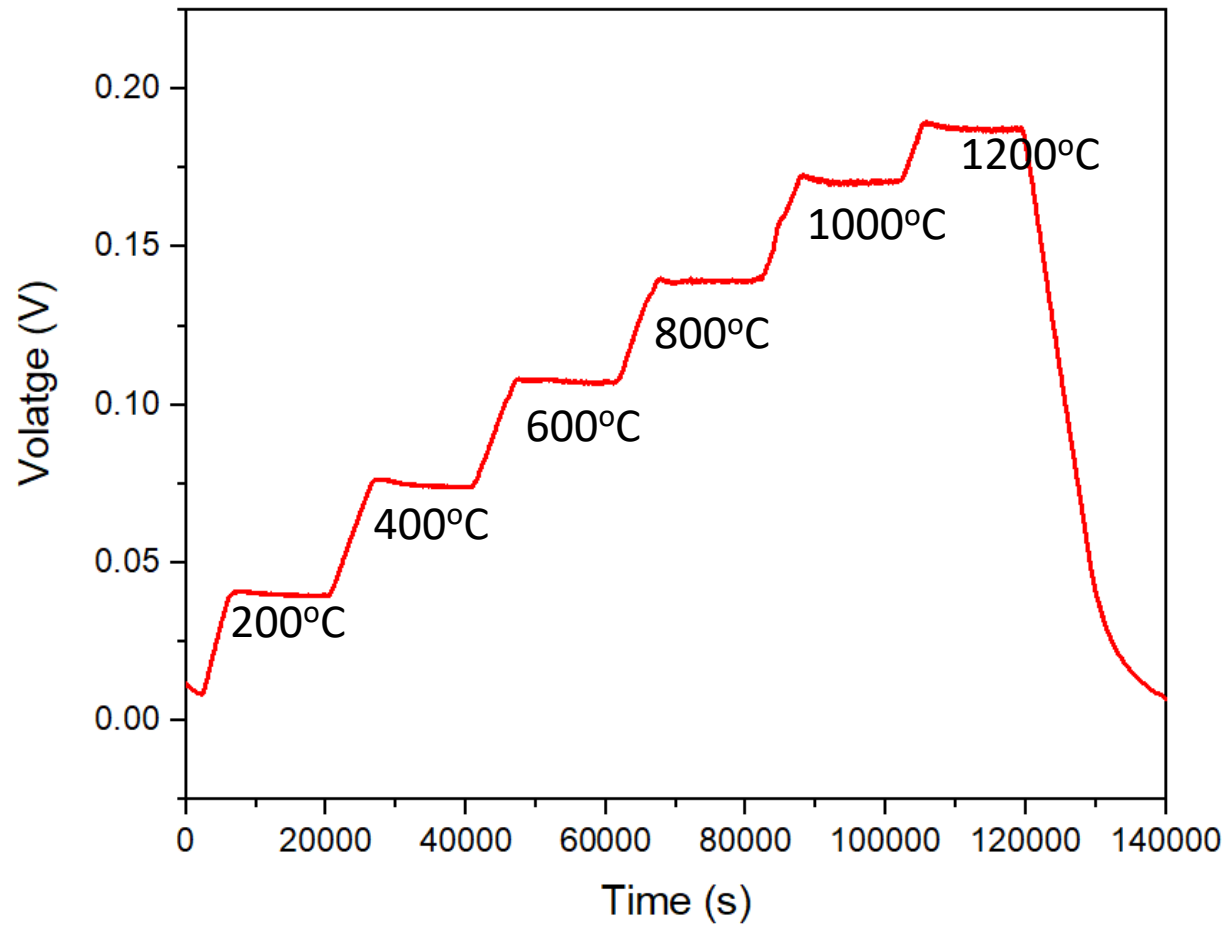


**1 Alumina layer**  
**1 LCC/ Alumina 95/5 layer**  
**1 Alumina layer**

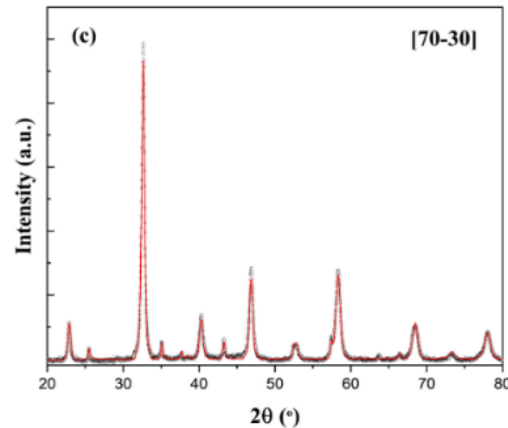
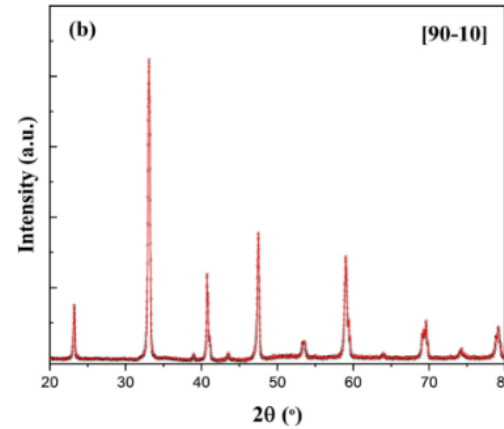
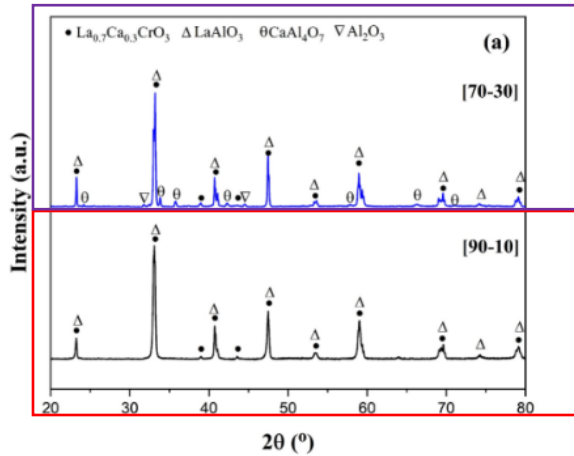




## 2 layers LCC20 2 Layers LC

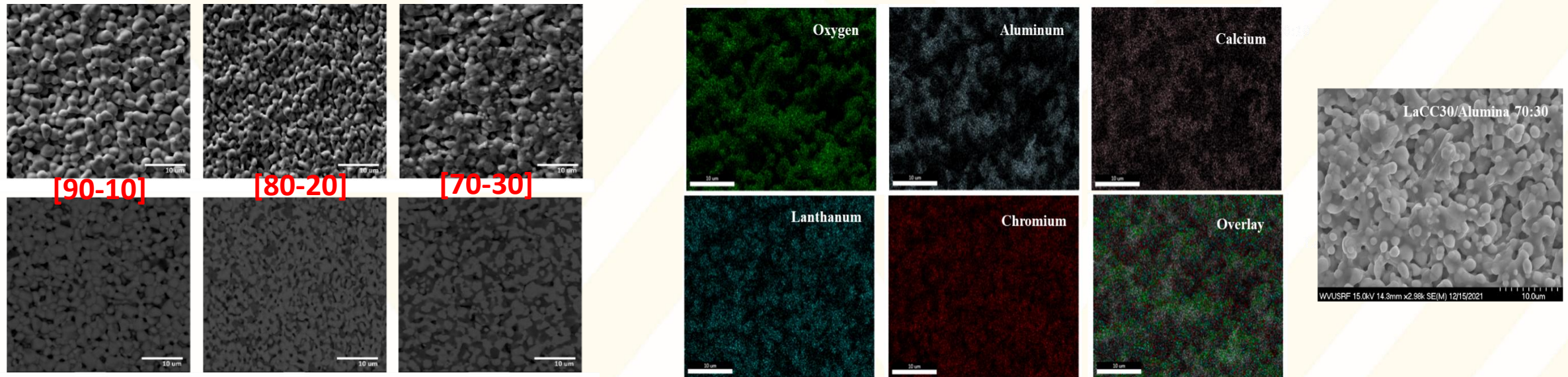


# XRD - Composites characterization



- ❖ LCC30-Al<sub>2</sub>O<sub>3</sub> [90-10] indicates that LCC30 phase was present, and a secondary lanthanum aluminate was formed during the sintering process.
- ❖ LCC30-Al<sub>2</sub>O<sub>3</sub> [70-30] diffractogram shows LCC30 and lanthanum aluminate (LaAlO<sub>3</sub>), calcium aluminate (Ca<sub>2</sub>Al<sub>2</sub>O<sub>5</sub>) and pure Al<sub>2</sub>O<sub>3</sub>.
- ❖ Increasing the Al<sub>2</sub>O<sub>3</sub> volume content in the composite, increase the chemical reactivity at 1500°C.

# SEM - Composites characterization



- ❖ At 10 vol% of  $\text{Al}_2\text{O}_3$  there is not connectivity between grains indicating the formation of a (3-0) composite.
- ❖ At 20 vol% and 30 vol% the connectivity of the grains is more notable, indicating the formation of (3-3) composites.
- ❖ These results demonstrated that the degree of percolation of  $\text{Al}_2\text{O}_3$ ,  $\text{LaAlO}_3$  and  $\text{Ca}_2\text{Al}_2\text{O}_5$  grain size increased by increasing  $\text{Al}_2\text{O}_3$  content from 10 to 30 vol%.