

# High Temperature Thick Film Sensor Development Based on Refractory Oxide Semiconductors

Javier A. Mena<sup>1</sup>, Katarzyna Sabolsky<sup>1</sup>, Anthony A. Abrahamian<sup>1</sup>, Domenic T. Cipollone<sup>1</sup>,  
Konstantinos Sierros<sup>1</sup>, Edward M. Sabolsky<sup>1</sup>, Víctor Mendoza-Estrada<sup>2</sup>

<sup>1</sup>Department of Mechanical and Aerospace Engineering,  
West Virginia University, Morgantown, WV 26506, USA

<sup>2</sup>Applied Physics Research Group, Physics Department,  
Universidad del Norte, Barranquilla, Colombia

**ICMSN- 22**  
**London, UK**  
**July 12-14, 2022**

# Introduction

---

- Processes such as energy generation, metals/glass manufacturing, coal gasification and aerospace technology applications require health and process monitoring in harsh-environments.
- **Harsh-environments conditions include:**
  - ❖ High temperature (500-1800°C)
  - ❖ High pressure (up to 1000 psi)
  - ❖ Corrosive, erosive and reducing environments.
- **Ability to monitor:**
  - ❖ Temperature
  - ❖ Structural stability of systems components
- **US DOE Overall Goal:** Develop health and temperature sensors (and sensor arrays) printed on refractory produced by advanced 2D/3D printing.



# *Objectives of This Work*

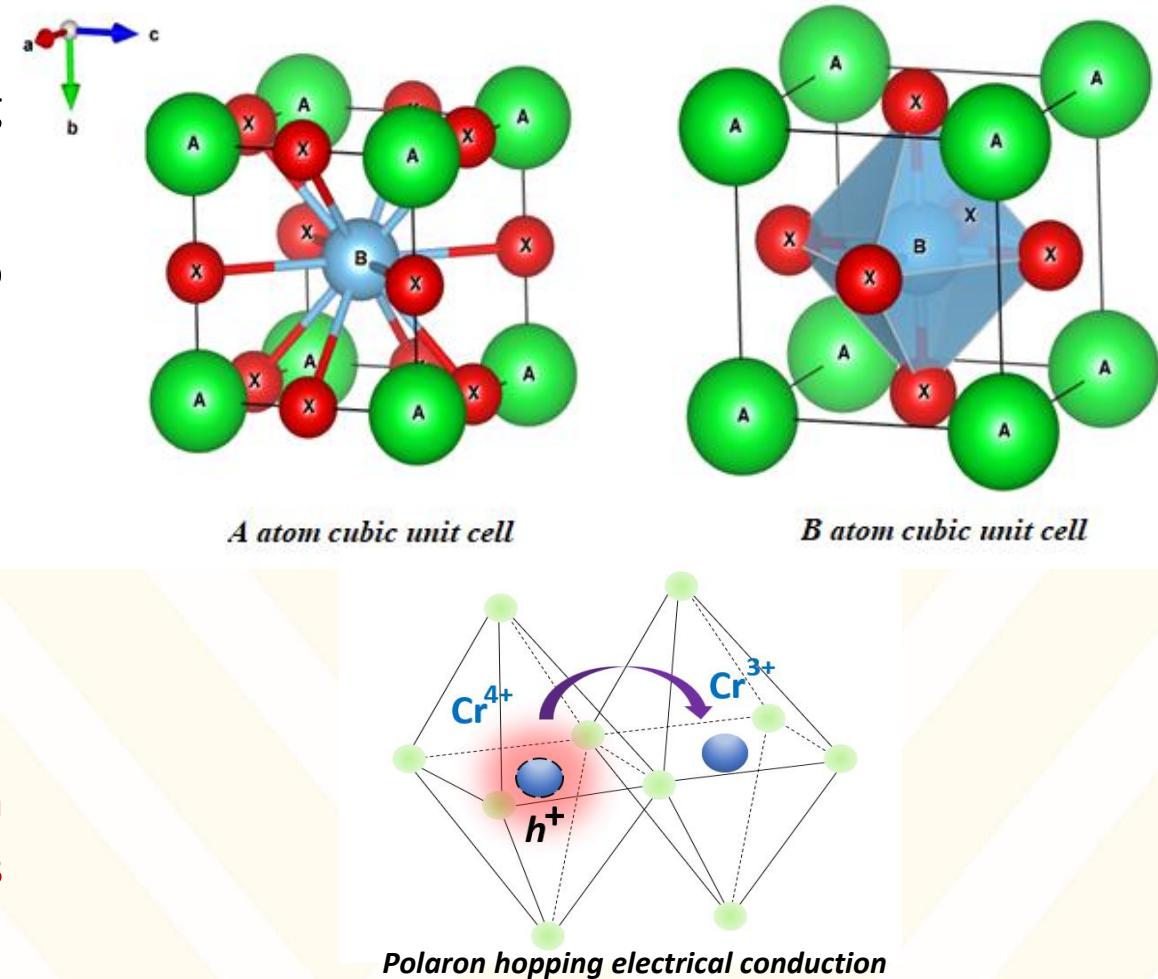
---

- 1) Synthesize high-temperature refractory  $\text{LaCrO}_3$  powders with various p-type doping schemes using the Pechini sol-gel method.
- 2) Characterization phase development/structure (XRD), microstructure and grain size distribution (SEM), optical properties (UV-Vis) and bulk density.
- 3) Study the electrical properties (electrical conductivity and Seebeck coefficient) of compositions at high temperatures conditions and various atmospheres (oxidizing, reducing, different oxygen partial pressures) of these compositions.
- 4) Computational DFT modeling of the  $\text{LaCrO}_3$  structure and electrical properties.
- 5) Fabricate surface printed thick-film temperature sensors utilizing these materials and test at high-temperature.

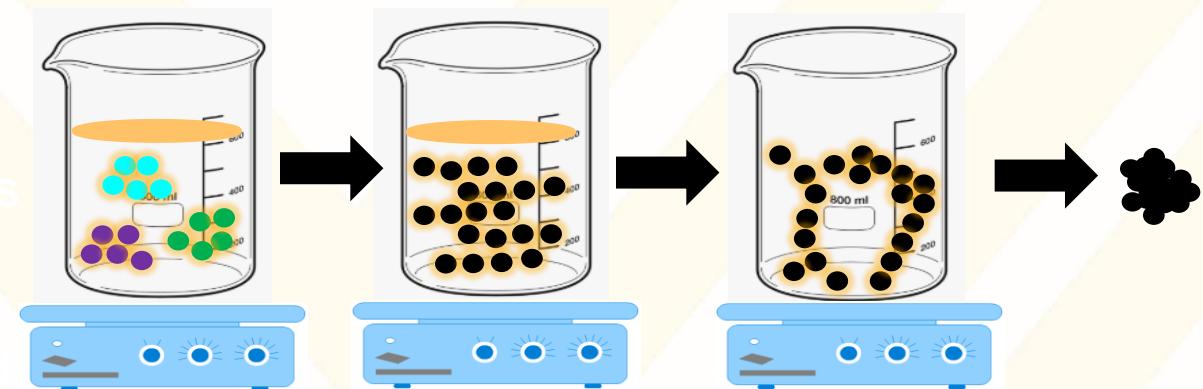
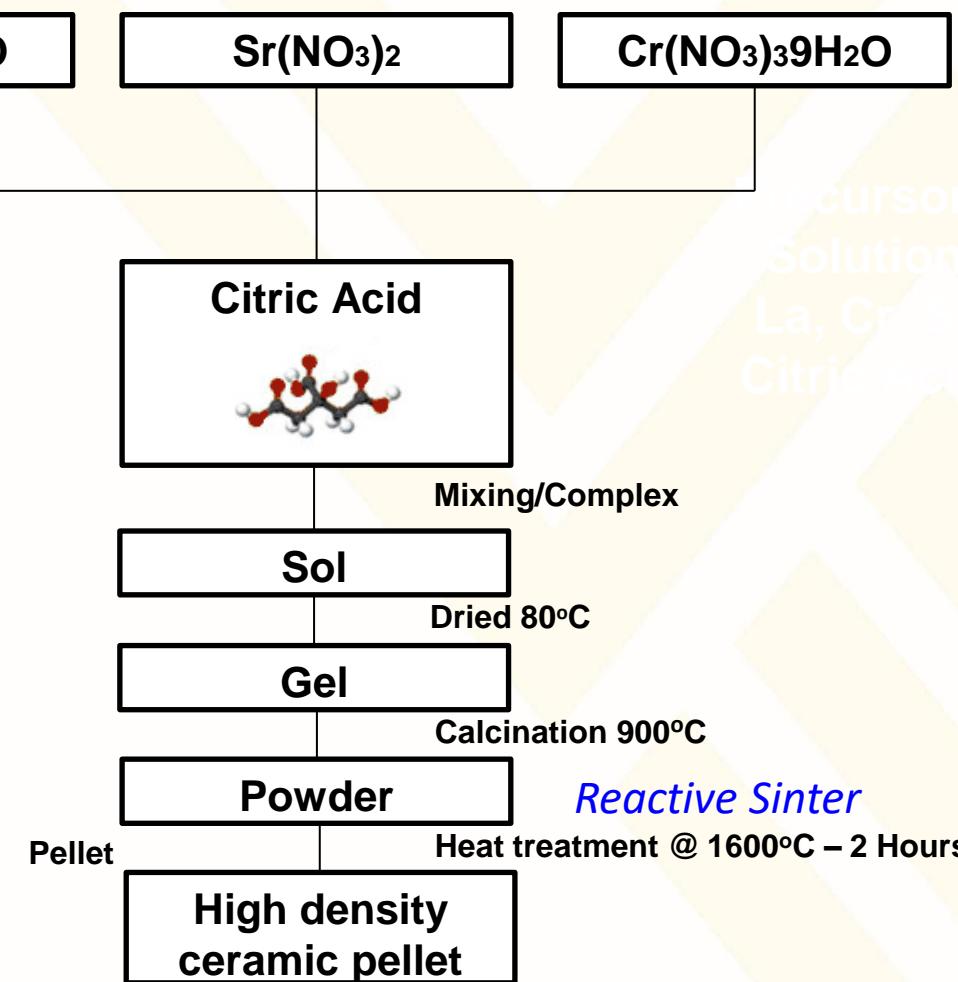
# Lanthanum Chromite: General Aspects

- High melting point ( $\sim 2500$  °C).
- Chemically stable under oxidative and reducing atmospheres.
- Pure  $\text{LaCrO}_3$  shows semiconducting behavior with no to low ionic conduction.
- $\sigma = >10-100 \text{ S}\cdot\text{cm}^{-1}$  (RT – 1000°C).
- Compatibility (thermal expansion coefficients matching) near refractory materials,  $\sim 10 \times 10^{-6} \text{ C}^{-1}$ .

\*Electrical properties (mostly in the 1990-2000's) in literature typically shown for  $<1000^\circ\text{C}$  due to original focus for Solid-Oxide Fuel Cell (SOFC) applications.



# *Sol Gel Synthesis and Pellet Fabrication*



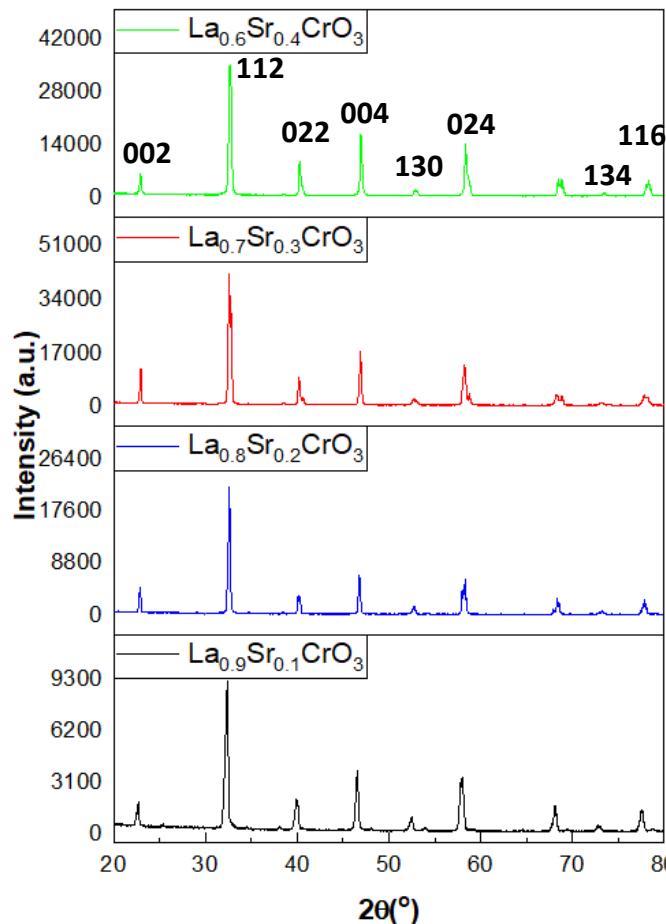
- ✓ Pechini-like process used.
- ✓ High homogenous and adequate sintering.
- ✓ High density (typical in literature  $<93\%$  density).
- ✓ Low yields and not easy to scale-up

## Compositions Studied:

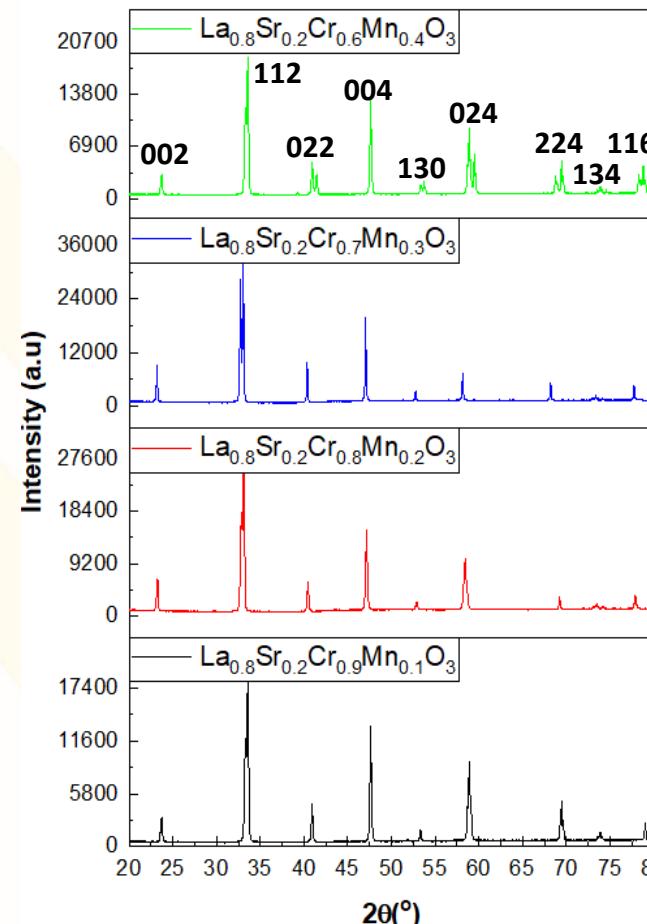
**A-site:**  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$   
( $x = 0, 0.1, 0.2, 0.4$ )

**B-site:**  $\text{La}_{0.8}\text{Sr}_{0.2}\text{Cr}_{1-y}\text{Mn}_y\text{O}_3$   
( $y = 0, 0.1, 0.2, 0.4$ )

# Crystalline Structure/Phase Analysis



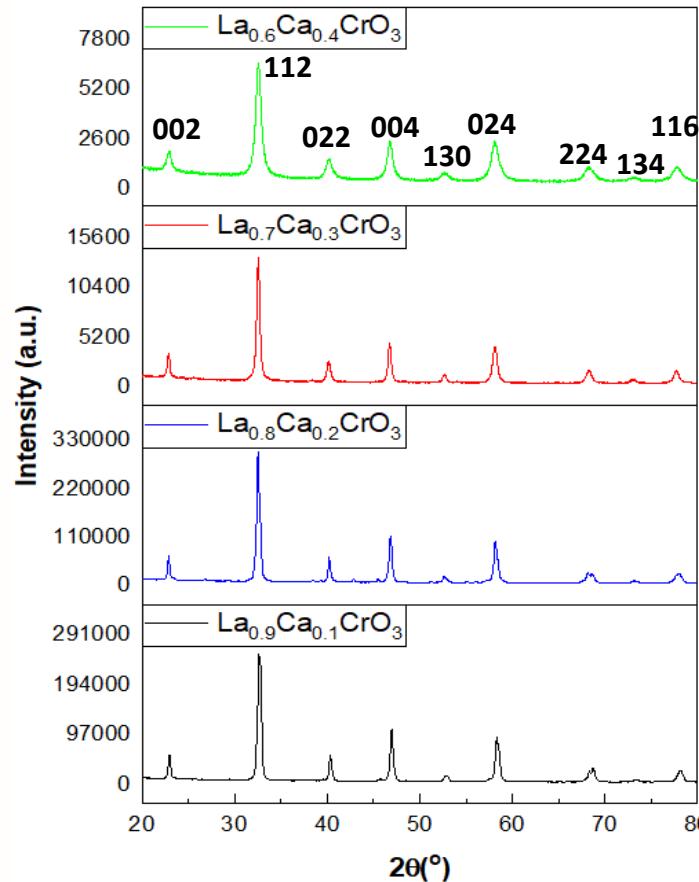
X-ray diffractograms for the samples of the  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$  series



X-ray diffractograms for the samples of the  $\text{La}_{1-x}\text{Sr}_x\text{Cr}_{1-y}\text{Mn}_y\text{O}_3$  series

- ❖ Single phase doped lanthanum chromites materials were obtained successful (no residual oxide or pyrochlore peaks).
- ❖ Using Pechini Sol-Gel method permitted doped lanthanum chromites at high solubility levels (40%).
- ❖ No impurities extra peaks were present in the final prepared powders and ceramic pellets.

# Lattice Parameters, Unit Cell Volume and Density



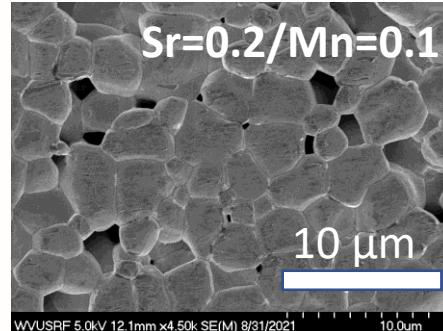
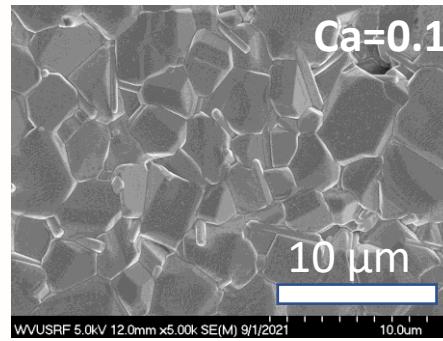
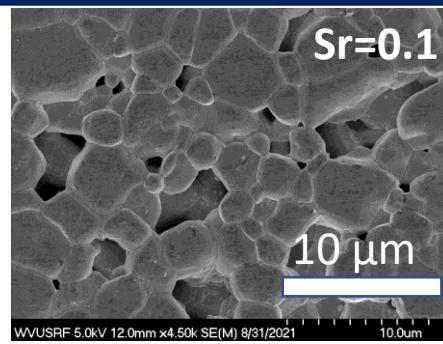
X-ray diffractograms for the samples of the  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$  series

Lattice parameters, unit cell volume and XRD theoretical density for doped lanthanum chromites perovskites

Composition	Lattice parameters (Å)			Volume (Å <sup>3</sup> )	$\rho_{XRD\ Theoretical}$ (g/cm <sup>3</sup> )
	a	b	c		
$\text{La}_{0.9}\text{Sr}_{0.1}\text{CrO}_3$	5.5124	5.5668	7.7926	239.1299	6.4932
$\text{La}_{0.8}\text{Sr}_{0.2}\text{CrO}_3$	5.4988	5.5425	7.7853	237.2747	6.4004
$\text{La}_{0.7}\text{Sr}_{0.3}\text{CrO}_3$	5.4769	5.5233	7.7580	234.6839	6.3259
$\text{La}_{0.6}\text{Sr}_{0.4}\text{CrO}_3$	5.4524	5.5122	7.7407	232.6441	6.2350
$\text{La}_{0.9}\text{Ca}_{0.1}\text{CrO}_3$	5.4180	5.5039	7.7332	230.6050	6.5963
$\text{La}_{0.8}\text{Ca}_{0.2}\text{CrO}_3$	5.4092	5.4982	7.7264	229.7898	6.3341
$\text{La}_{0.7}\text{Ca}_{0.3}\text{CrO}_3$	5.3994	5.4877	7.7058	228.3264	6.0872
$\text{La}_{0.6}\text{Ca}_{0.4}\text{CrO}_3$	5.3897	5.4622	7.6853	226.2520	5.8528
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.90}\text{Mn}_{0.10}\text{O}_3$	5.4734	5.5648	7.7765	236.8595	6.1533
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.80}\text{Mn}_{0.20}\text{O}_3$	5.4705	5.5587	7.7702	236.2829	6.1766
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.70}\text{Mn}_{0.30}\text{O}_3$	5.4598	5.5398	7.7498	234.4020	6.2344
$\text{La}_{0.8}\text{Sr}_{0.20}\text{Cr}_{0.60}\text{Mn}_{0.40}\text{O}_3$	5.4146	5.4981	7.7065	229.4225	6.3783

- ❖ Decrease in lattice parameters (and volume) were observed when dopant cations is introduced in the lattice.
- ❖ To achieve neutrality chromium oxidation states, change from  $\text{Cr}^{+3}$  to  $\text{Cr}^{+4}$ , reduction in the chromium ionic size occurs.

# Microstructure/Grain Size Distribution



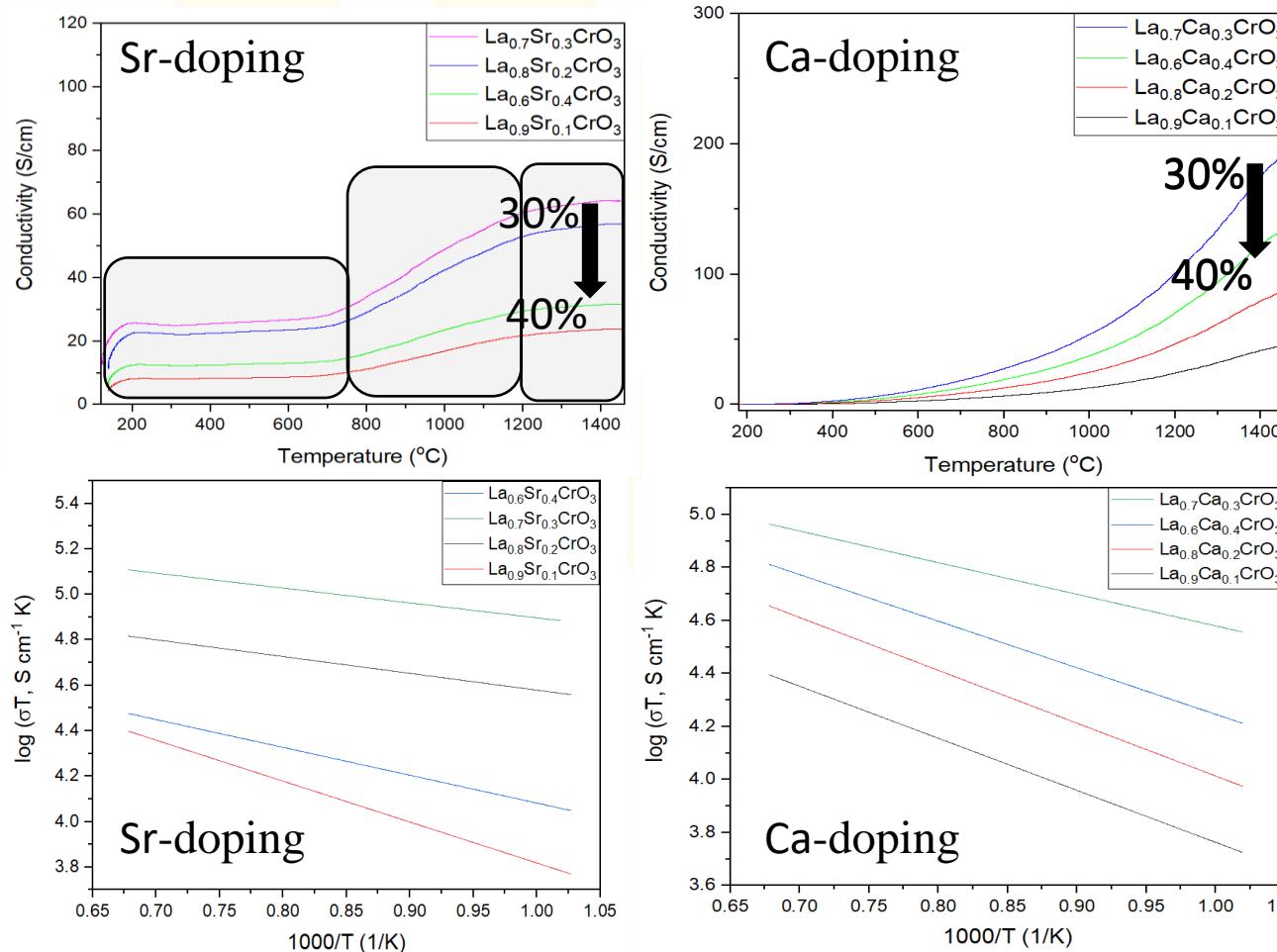
Average grain size and bulk density distribution for  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Sr}_x\text{Cr}_{1-y}\text{Mn}_y\text{O}_3$  series

Composition	Average Grain Size ( $\mu\text{m}$ )	Relative Percentage Bulk Density (%)
$\text{La}_{0.9}\text{Sr}_{0.1}\text{CrO}_3$	3.6480	94
$\text{La}_{0.8}\text{Sr}_{0.2}\text{CrO}_3$	3.4898	95
$\text{La}_{0.7}\text{Sr}_{0.3}\text{CrO}_3$	3.6290	95
$\text{La}_{0.6}\text{Sr}_{0.4}\text{CrO}_3$	3.2455	94
$\text{La}_{0.9}\text{Ca}_{0.1}\text{CrO}_3$	4.1367	96
$\text{La}_{0.8}\text{Ca}_{0.2}\text{CrO}_3$	3.7727	97
$\text{La}_{0.7}\text{Ca}_{0.3}\text{CrO}_3$	3.6883	97
$\text{La}_{0.6}\text{Ca}_{0.4}\text{CrO}_3$	3.6548	98

- ❖ Pechini Sol Gel prepared calcium, strontium, manganese doped lanthanum chromite powders exhibit better sinterability and densification under oxidizing conditions (undoped  $\downarrow 90\%$ ).
- ❖ The samples of Ca doped lanthanum chromite powder have more dense microstructures. Furthermore, it was found that the incorporation of Ca, Sr in the A site of the lanthanum chromite increases grain growth (undoped  $< 3 \mu\text{m}$ ).



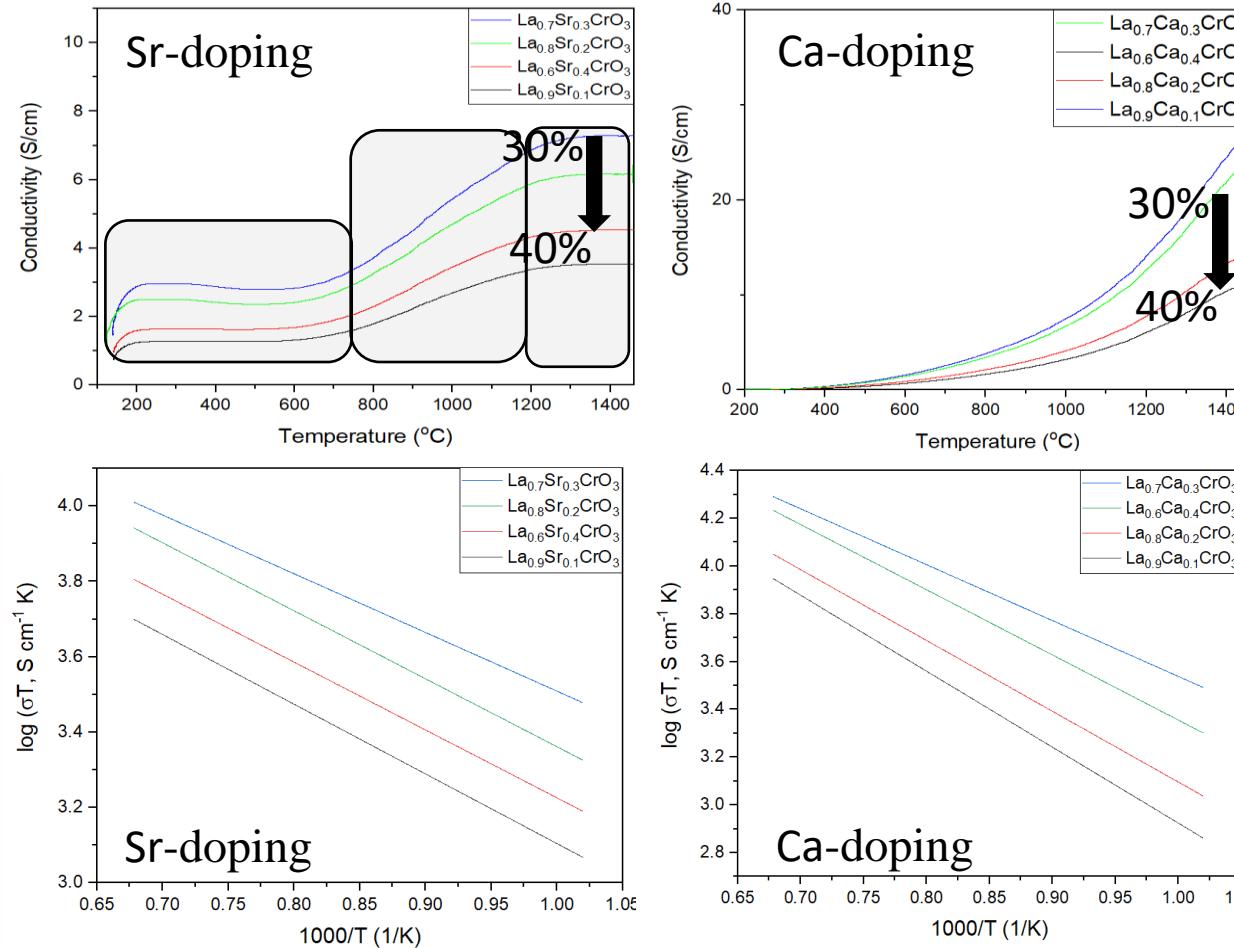
# DC Electrical Conductivity (Oxidizing Atmosphere)



Electrical conductivity vs temperature and Arrhenius Plot  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$

- ❖ Conductivity typically exponentially increases with increase in carrier mobility, but 30 to 40% all drop in conductivity (believe slight second phase or higher lattice strain).
- ❖ Sr doped shows three regions, not seen in literature, since most tests  $<850\text{--}1000\text{ }^\circ\text{C}$ . (believe  $V_\text{o}$  at high temperature)
- ❖ Arrhenius relationship fits for higher temperature regimes.
- ❖ Calcium doped compositions present higher conductivity due the lower distortion effects on lattice structure.

# DC Electrical Conductivity (Reducing Atmosphere)



Electrical conductivity vs temperature and Arrhenius Plot for  $\text{La}_{1-x}\text{Sr}_x\text{CrO}_3$ ,  $\text{La}_{1-x}\text{Ca}_x\text{CrO}_3$

- ❖ Conductivity decrease for all temperature range under reducing atmosphere (H<sub>2</sub> 5% / N<sub>2</sub> 95%).
- ❖ Under reducing conditions, oxygen vacancies form to keep neutrality (drop in hole carrier concentration).
- ❖ Sr conductivity still displays regions of altered mechanism.
- ❖ All compositions present lower conductivity as dopant increases (but still 40% lower than 30% in all cases).

# DC Electrical Conductivities/Activation Energies

Composition	Air Atmosphere		Reducing Atmosphere	
	Conductivity @ 1000°C (S/cm)	Activation energy (eV)	Conductivity @ 1000°C (S/cm)	Activation energy (eV)
La <sub>0.9</sub> Sr <sub>0.1</sub> CrO <sub>3</sub>	16.672	0.1552	2.691	0.3238
La <sub>0.8</sub> Sr <sub>0.2</sub> CrO <sub>3</sub>	42.882	0.1427	4.572	0.2597
La <sub>0.7</sub> Sr <sub>0.3</sub> CrO <sub>3</sub>	49.032	0.1055	5.534	0.1719
La <sub>0.6</sub> Sr <sub>0.4</sub> CrO <sub>3</sub>	23.599	0.1498	3.462	0.3102
La <sub>0.9</sub> Ca <sub>0.1</sub> CrO <sub>3</sub>	13.211	0.1417	3.152	0.3240
La <sub>0.8</sub> Ca <sub>0.2</sub> CrO <sub>3</sub>	24.170	0.1298	4.118	0.2103
La <sub>0.7</sub> Ca <sub>0.3</sub> CrO <sub>3</sub>	52.823	0.1028	7.602	0.1367
La <sub>0.6</sub> Ca <sub>0.4</sub> CrO <sub>3</sub>	38.152	0.1175	6.626	0.1747

$$\sigma = \frac{\sigma_0}{T} \exp\left(\frac{-\Delta E_a}{kT}\right)$$



Slope  $\propto E_a$

- ❖ Conductivity increase as function of doping level up to 30% for all dopants (strontium, calcium and manganese).
- ❖ At 40% doping levels conductivity decrease due to higher lattice distortion in all systems (solubility limit).
- ❖ Calcium and strontium/manganese doping obeys Arrhenius exponential law.
- ❖ Lower conductivity values under reducing atmospheres are explained by oxygen vacancies formation (near  $>1.5\times$  in activation energy).

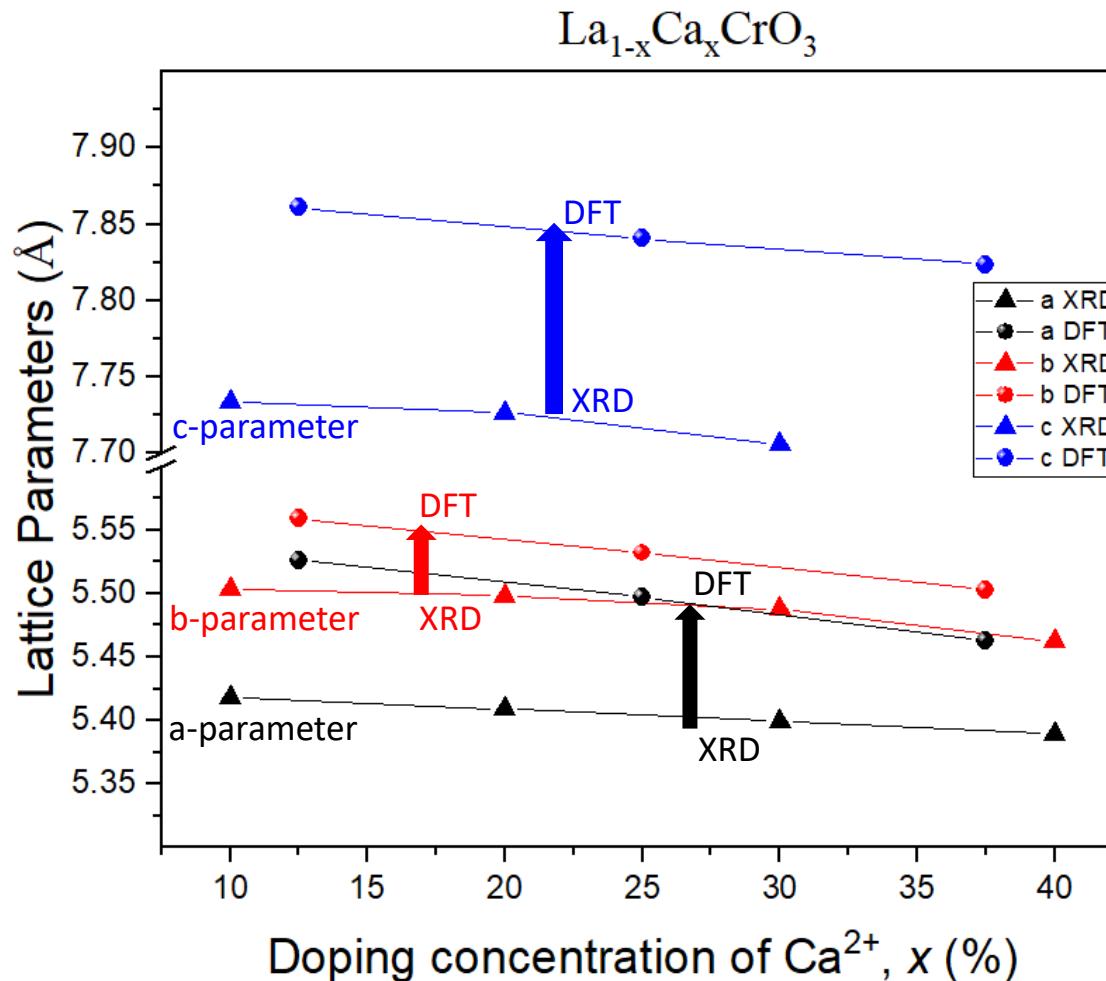
# **Computational Modelling for Ca-Doping Compositions**

---

- ❖ First-principles spin-polarized calculation within the density functional theory (DFT) framework were performed.
- ❖ Exchange and correlation effects were treated with generalized gradient approximation (GGA) and GGA+U method implemented in the Perdew-Burke-Ernzerhof functional (PBE). Vienna Ab initio Simulation Package (VASP) was used.
- ❖ A cutoff energy of 520 eV, and a  $3 \times 6 \times 4$  k-point mesh was employed for a  $2a \times 1b \times 1c$   $LaCrO_3$  orthorhombic supercell.
- ❖ The transport properties such as electrical conductivity ( $\sigma$ ) was calculated using the BoltzTraP2 code that is formulated according to classical Boltzmann transport equation under the constant relaxation time ( $\tau$ ) approximation ( $\sigma$  can only be obtained in units of  $\tau$ , since there is no experimental data available on the relaxation time in this material).

*Work by: Victor Julio Mendoza Estrada (at Universidad del Norte)*  
*Email: evictor@uninorte.edu.co*

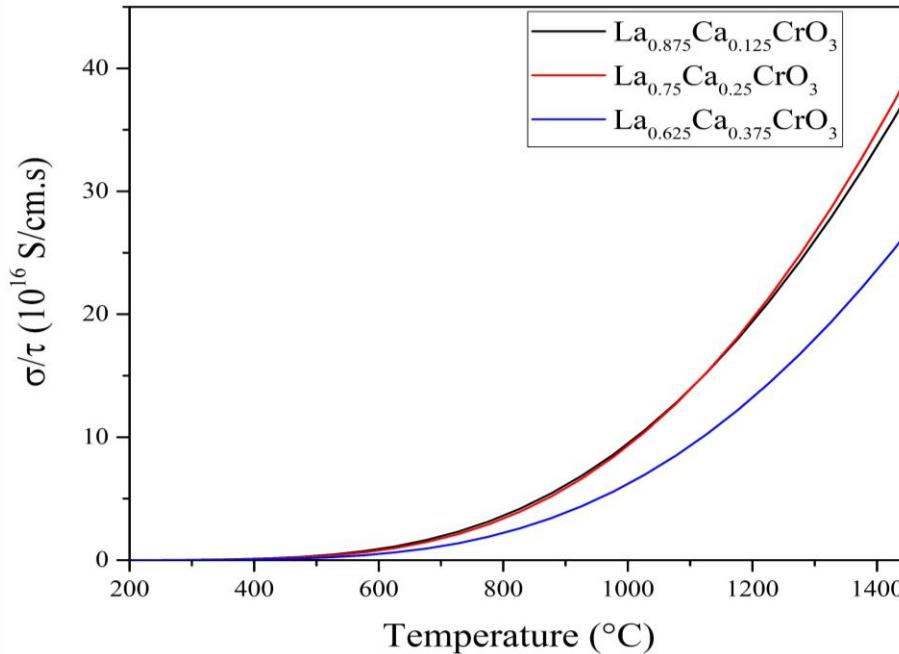
# DFT and Experimental Lattice Parameters



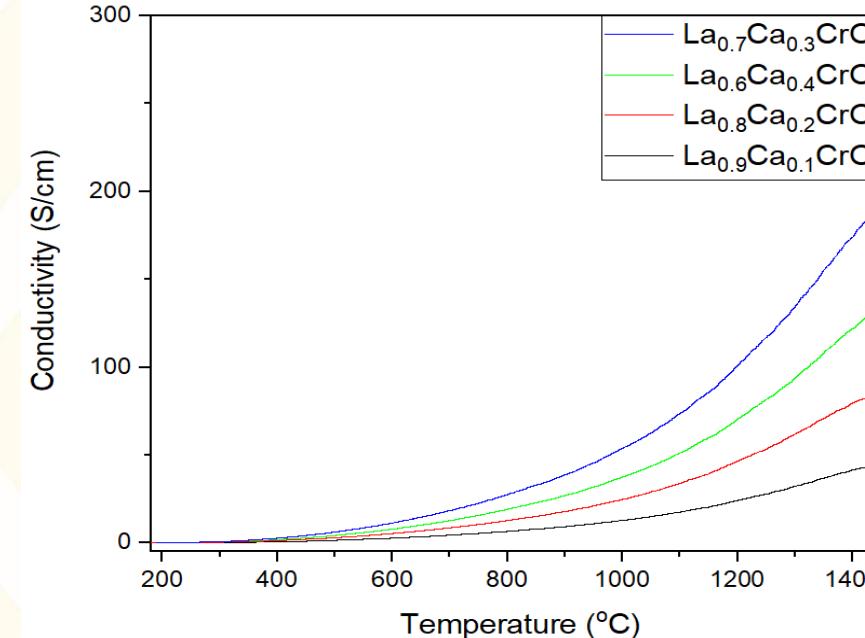
- ❖ Decrease in lattice parameters were observed when dopant cations are introduced in the lattice.
- ❖ To achieve neutrality chromium oxidation states, change from  $\text{Cr}^{+3}$  to  $\text{Cr}^{+4}$ , reduction in the chromium ionic size occurs.
- ❖ DFT theoretical data shows equal trends in the correlation between dopant level and lattice parameters as experimental data.
- ❖ DFT calculations of lattice parameters over-estimates lattice parameters values for all dopants and compositions in contrast with experimental XRD measurements.

# Calculated Electrical Conductivity for Ca-Doping

DFT Results



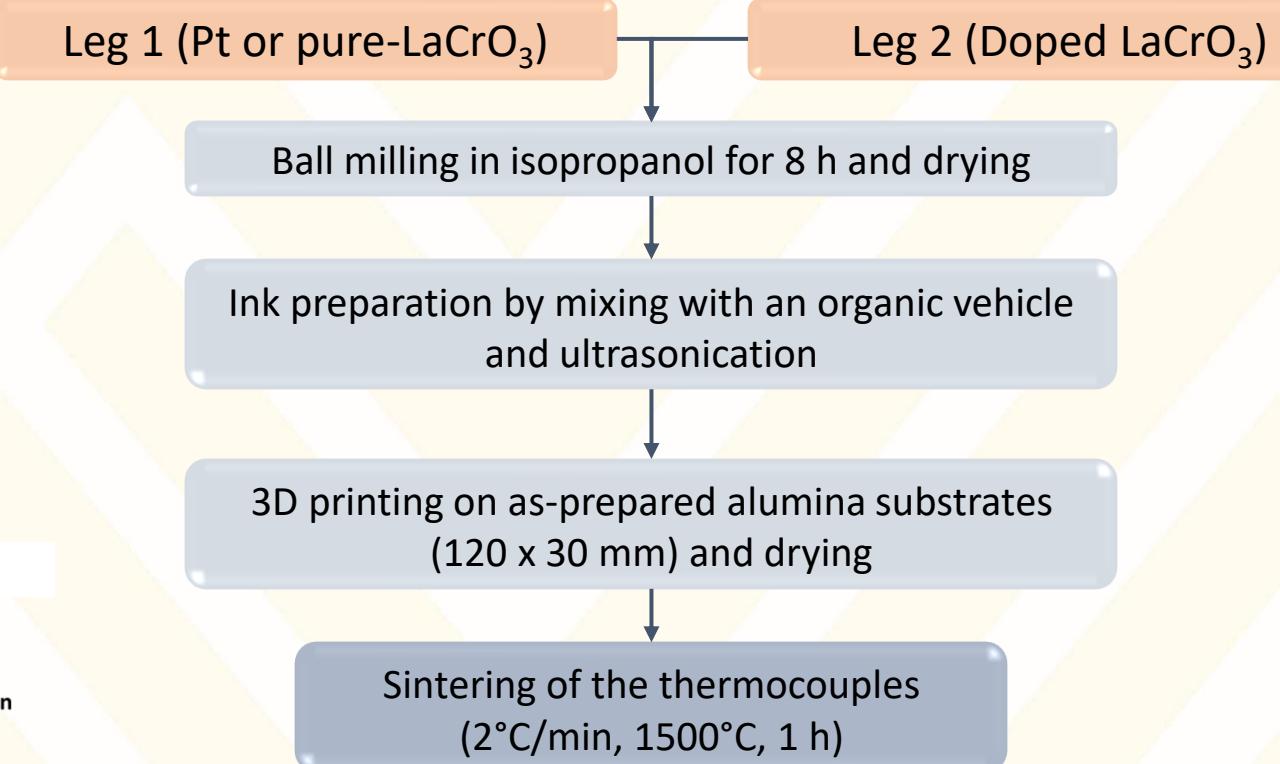
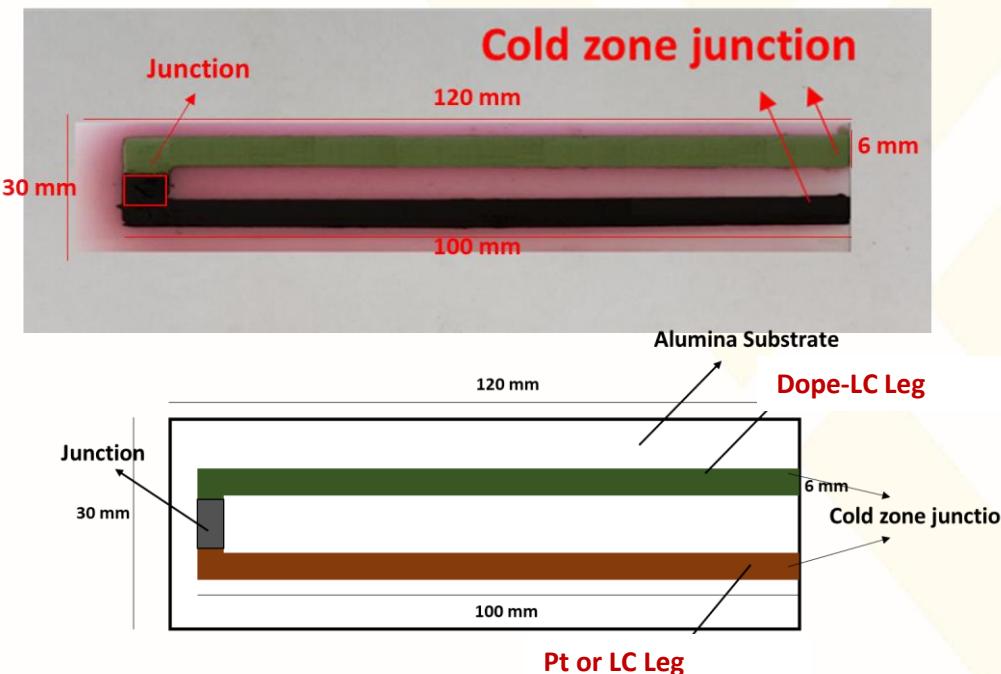
Experimental Results



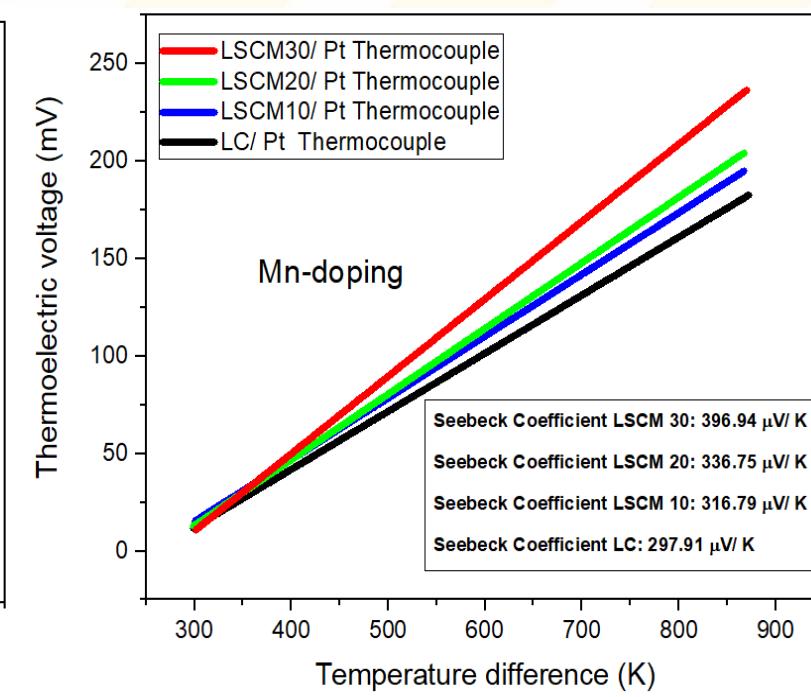
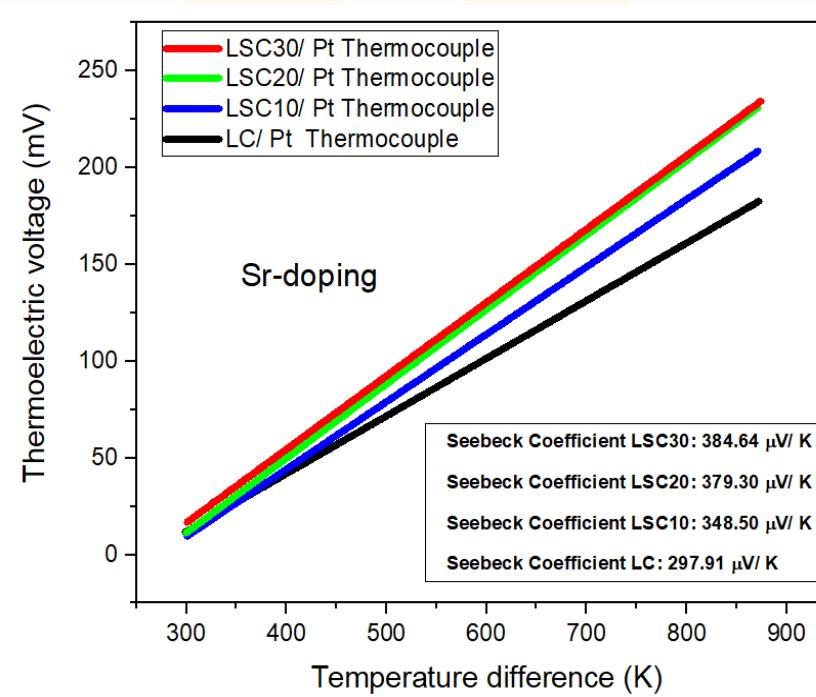
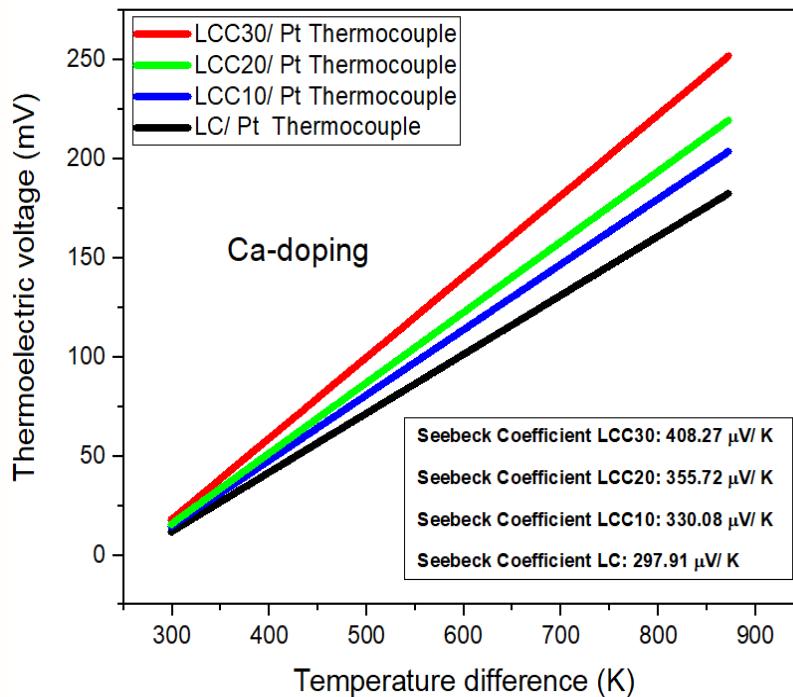
- ❖ The Ca-doped electrical conductivity presents proportional correlation increment with temperature, which is agreement with our experimental results.
- ❖ DFT calculated conductivity increase as a function of doping level up to 25% for calcium doping.
- ❖ DFT differ from experimental where 25-37.5% show similar conductivity, but not shown for experimental.

# High Temperature Thermocouple Fabrication

- Research team currently fabricating strain, spallation, failure, thermistors and thermocouples by Direct Ink Writing (DIW) 3D printing.
- High-temperature thermocouples that function >1200°C (in R-type range) new exciting development.



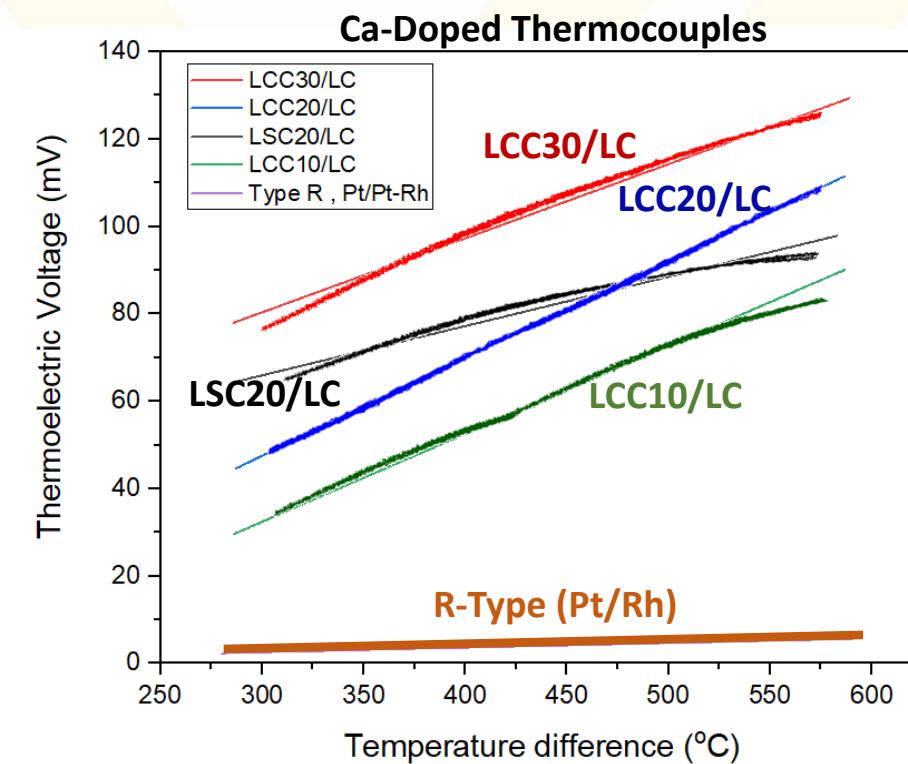
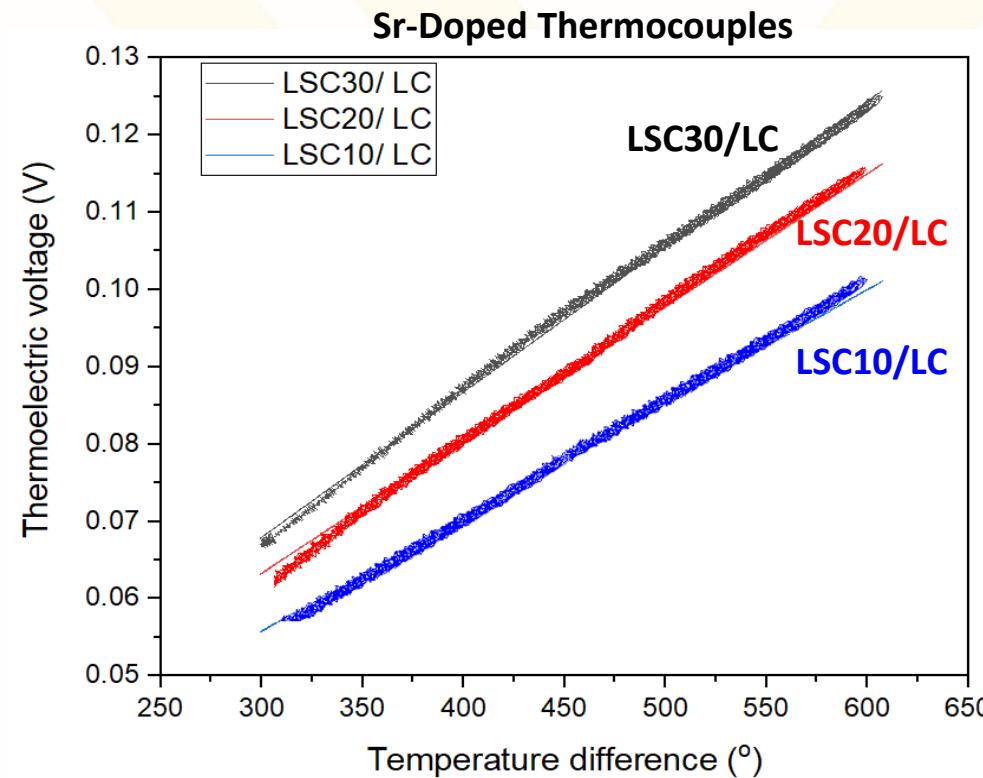
# Seebeck Coefficient Estimation (Using Pt Standard)



- Linear correlation between temperature difference and thermoelectric voltage was observed for all the compositions.
- Doped- $\text{LaCrO}_3$ /Pt couples were fabricated to estimate absolute Seebeck coefficient ( $S_{\text{Pt}} \sim -18 \mu\text{V}/\text{K}^*$ ) up to 1000°C.
- Ca doping shows highest absolute Seebeck coefficient (330-408  $\mu\text{V}/\text{K}$ ), with increasing Ca content.
- Ca doping on A-site has similar effect as Mn doping on B-site for LSC compositions.

\*Moore, J. P. (1973). Journal of Applied Physics. 44 (3): 1174–1178

# Thermoelectric Characterization of Thermocouples



- Thermocouples were tested in a range between 30 to 750°C during 3 heating cycles, showing an excellent reproducibility.
- Sr-doped compositions have lower Seebeck coefficient over Ca-doped compositions.
- The LCC30/LC, LCC20/LC and LCC10/LC thermocouples showed a maximum higher voltage by two orders of magnitude in comparison with Pt/Pt-Rh, with values of 124.61 mV, 116.50 mV and 80 mV respectively (at  $\Delta T \sim 550^\circ\text{C}$ ).

# Conclusions

---

- Using **Pechini Sol-Gel** method approach **high density (~96%)** doped lanthanum chromite compositions can be obtained. **Uniform phase** composition can be produced easily.
- **Lattice parameters reduce** proportionally with dopant content increment. **Divalent dopants produce** chromium oxidation state **changes** from +3 to +4 **reducing the ionic radius**.
- **Electrical conductivity** shows correlated **dependence** with high **temperature (up to 1500°C)** for all compositions. The exponential Arrhenius trend is evidence for the **polaron hopping electrical conduction** mechanism.

# Conclusions

---

- Conductivity increase in function of doping level up to 30% for all dopants (strontium, calcium and manganese).
- Under reducing conditions conductivity decrease, chromium changes oxidation state from +4 to +3 and oxygen vacancies form to keep neutrality.
- Doping increases Seebeck coefficient and permits functional thermocouples to be fabricated with pure  $\text{LaCrO}_3$  (showing thermoelectric voltages 2x greater than Pt thermocouples).

# Acknowledgment

---

- ❖ We would like to thank **U.S. Department of Energy (DOE)** for sanctioning this project **DE-FE0031825**.
- ❖ We also would like to acknowledge Mr. Harley Hart, Dr. Qiang Wang, and Dr. Marcela Redigolo for their cooperation and valuable assistance in the WVU Shared Facilities.
- ❖ We also would like to thank HWI, for support us in developing real-life applications sensing systems/devices.
- ❖ Kindly acknowledge faculty and staff of West Virginia University for their support.



***Thank you for the attention.***

***Email:***

***ed.sabolsky@mail.wvu.edu***