

# First-Principles Studies of Tritium Species Diffusivity Across the Interface of Nickel- Plated Zircaloy-4



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***2024 9<sup>th</sup> Annual Tritium Science Technical Exchange***

***Sept. 17, 2024***



# Disclaimer

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# Scope of Work

## Three Subtasks

- **Task 1:** Exploring  $^3\text{H}$  diffusion across pure Ni-Zr interface. The diffusion pathway will be obtained and serves as our baseline for the following tasks. **Completed**
- **Task 2:** Calculating the  $^3\text{H}$  diffusion pathways across the Ni-Zr interface when  $\text{NiO}_x/\text{Ni}(\text{OH})_x$  appeared in the Ni side of the interface region. Based on the pure Ni-Zr interface model from Task 1, we will introduce  $\text{NiO}_x/\text{Ni}(\text{OH})_x$  clusters on the Ni side closer to the interface. Then, the  $^3\text{H}$  diffusion pathways and barriers from Ni side to the zircaloy-4 will be explored. We will investigate how  $\text{NiO}_x/\text{Ni}(\text{OH})_x$  affects the  $^3\text{H}$  diffusion across the interface. **Completed**
- **Task 3:** Exploring the effects of impurities on  $^3\text{H}$  diffusion in Ni-plated Zircaloy-4 getter. Based on the model constructed in Task 2, we will introduce impurities (e.g., Sn, Fe, C, defects) on the Zr side of the interface. Then, the pathways and barriers of  $^3\text{H}$  diffusion from Ni side to the zircaloy-4 will be calculated. From this task, we will determine whether the impurity can help or hinder the  $^3\text{H}$  diffusion into Zircaloy-4 getter to form  $\text{Zr}^3\text{H}_x$  phases for  $^3\text{H}$  storage. **Completed**

**NETL's supercomputer was shutdown from February through June.**



# Milestones/Deliverables

Tasks & Milestones	Schedules (Months)										
	11	12	1	2	3	4	5	6	7	8	9
Task1: Exploring $^3\text{H}$ diffusion across pure Ni-Zr interface											
Milestone 1: Obtain $^3\text{H}$ diffusion pathway across Ni-Zr interface											
Task2: Calculating the $^3\text{H}$ diffusion pathways with $\text{NiO}_x/\text{Ni}(\text{OH})_x$ cluster in interface											
Milestone 2: The effects of $\text{NiO}_x/\text{Ni}(\text{OH})_x$ on $^3\text{H}$ diffusion pathways & barriers											
Task3: Exploring the effects of impurities on $^3\text{H}$ diffusion in Ni-plated Zircaloy-4 getter											
Milestone 3: The impurity effects on $^3\text{H}$ diffusion pathways & barriers											
Milestone 4: Wrap up the calculations into manuscript and submit final report											
M: monthly report; S: semi-year report; F: final report	M	M	M	M	M	S	M	M	M	M	F

## Publications

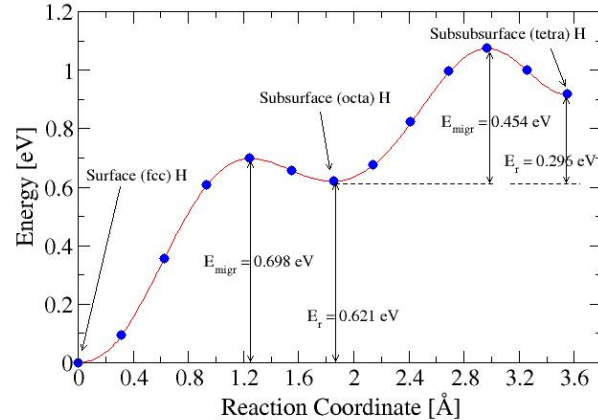
- (1) H. P. Paudel, T. Jia, W. A. Saidi, D. Senior, A. Casella, Y. Duan, "First-principles density functional study of tritium diffusion in alpha-Zirconium with C and Sn impurity," **Journal of Physical Chemistry C** **127**(26)(2023)12435-12443.
- (2) D. N. Tafen, Morgan Redington, H. P. Paudel, Y. Duan, "First-Principles Studies of Tritium Species Dissociability and Diffusivity Across the Interface of Nickel-Plated Zircaloy-4 Getters," **Tritium Science Program FY23 Final Report**, DOE/NETL-2023/3931, Dec. 2023. doi:10.2172/2279020.
- (3) D. N. Tafen, H. P. Paudel, D. J. Senior, A. M. Casella, Y. Duan, "Solubility and Diffusivity of Tritium Species in Interface of Nickel-Plated Zircaloy-4: First Principles Density Functional Study," **Phys. Chem. Chem. Phys.** (2024) submitted.
- (4) Morgan Redington, H. P. Paudel, D. N. Tafen, Daniel P. Miller, Eva Zurek, Y. Duan, "Tritium Adsorption and Absorption on (100) and (001) Surfaces of Pure and Tin Defective Zirconium," **J. Phys. Chem. C** (2024) to be submitted.
- (5) H. P. Paudel, D. N. Tafen, D. J. Senior, A. M. Casella, Y. Duan, "First-Principles Studies of Tritium Species Diffusivity Across the Interface of Nickel-Plated Zircaloy-4", (2024) under preparation.

## Presentations

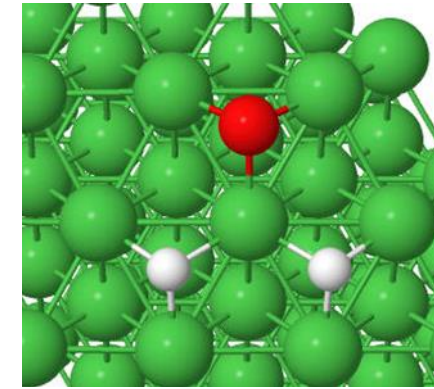
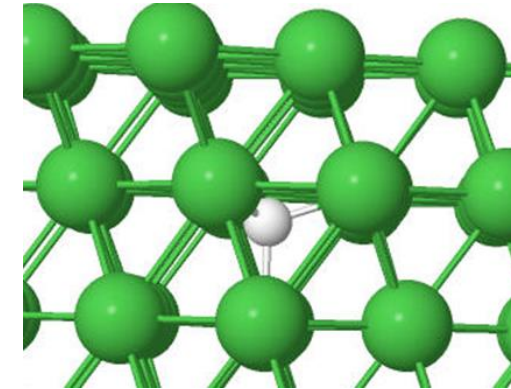
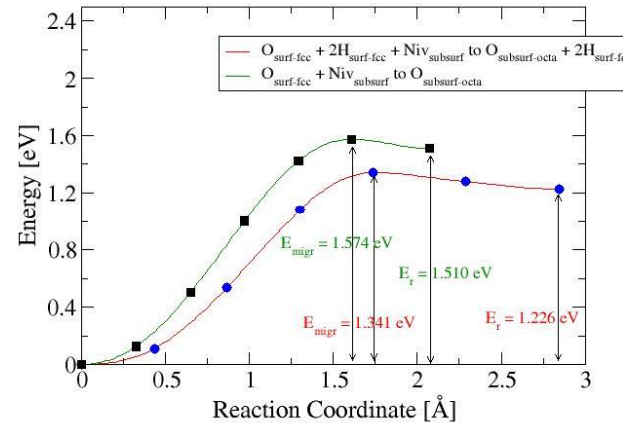
- (1) Y. Duan, H. Paudel, D. N. Tafen, T. Jia, D. Senior, A. M. Casella, "First-principles study of the tritium formation in  $\gamma$ - $\text{LiAlO}_2$  pellets and diffusion into Zircaloy-4 getter," **APS March Meeting**, Mar.03-08, 2024, Minneapolis, MN.

# Review from 2023: Tritium Diffusion in Ni (111) Surface

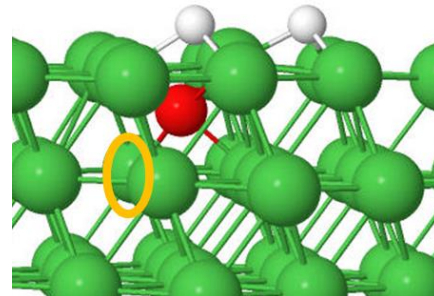
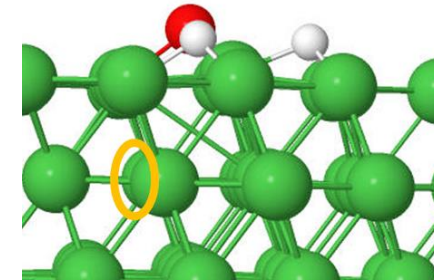
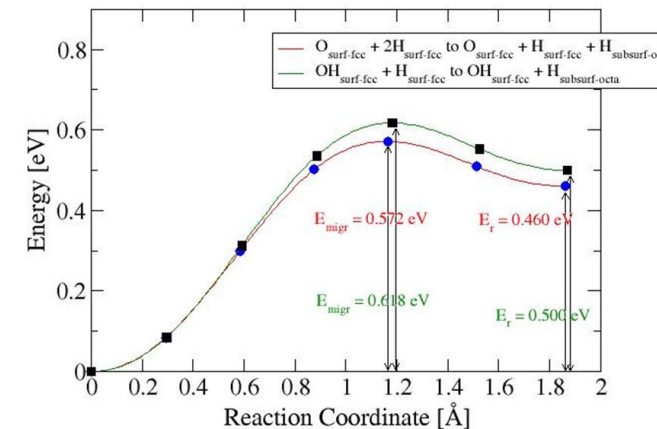
## Diffusion of H from Ni(111) surface to bulk



## H migration from surface to subsurface



- Explored  $^3\text{H}_2$  and  $^3\text{H}_2\text{O}$  binding sites and their dissociation on the (111) surface of  $^3\text{H}_2\text{O} \rightarrow \text{O}^3\text{H} + ^3\text{H} \rightarrow \text{O} + ^3\text{H} + ^3\text{H}$ , and  $^3\text{H}_2 \rightarrow ^3\text{H} + ^3\text{H}$ .
- Formation of  $\text{NiO}_x$  or  $\text{Ni}(\text{O}^3\text{H})_x$  and higher diffusion energy barrier for O as compared to that of  $^3\text{H}$ .
- Only  $^3\text{H}$  diffuses through Ni layer and reaches Ni-Zircalory-4 interface to form metal hydrides.

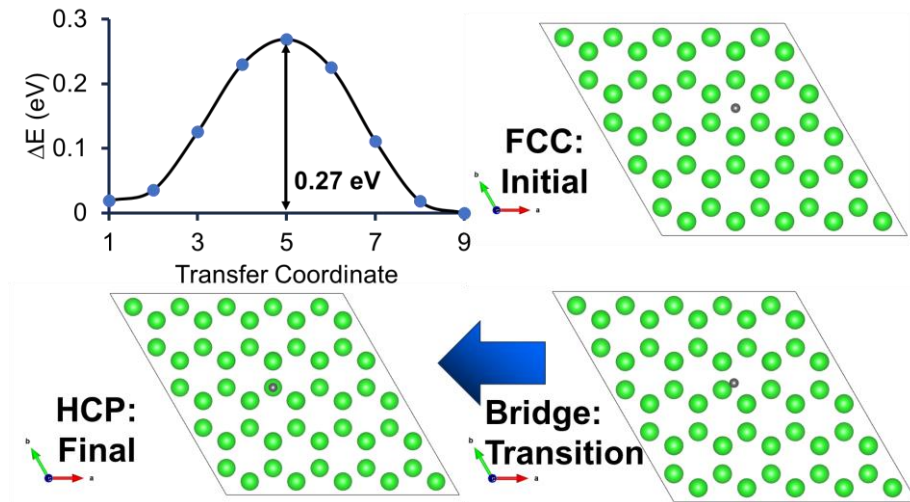


# Review from 2023: Tritium Diffusion in Zr (0001) Surface

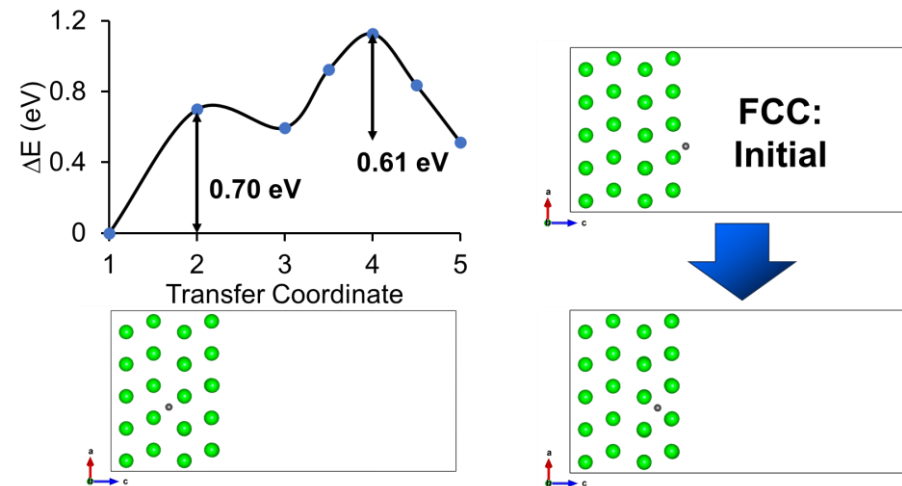
(Completed by Morgan Redington, SUNY Buffalo: 2023 Summer Intern)

The surface-surface and surface to subsurface diffusion barriers for  $^3\text{H}$  on these surfaces were calculated. The effect of Sn on  $^3\text{H}$  was shown to have a strong repulsive effect on these surfaces.

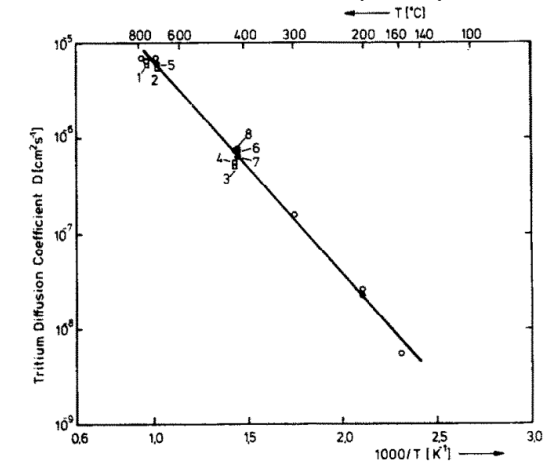
### Zr(100) Surface Transfer



### Zr(001) Surface to Bulk Transfer



*J. of Nuc. Mats. 88 (1980) 15-22*



$$D_0 = (1.04^{+0.28}_{-0.17}) \times 10^{-3} \quad [\text{cm}^2 \cdot \text{s}^{-1}] ,$$
$$E_A = -42.1 \pm 1.1 \quad [\text{kJ} \cdot \text{mol}^{-1}]$$
$$\sim 0.43 \text{ eV}$$

- The largest diffusion coefficient on the surface was found to be  $9.53 \times 10^{-10} \text{ m}^2/\text{s}$  at 600 K.
- A surface-subsurface diffusion coefficient of  $5.14 \times 10^{-13} \text{ m}^2/\text{s}$  at 600 K was calculated, with Sn impurities determined to reduce this by up to 9 orders of magnitude.



# Creating Zr/Ni Interface Models

## Lattice Parameters Mismatch Minimization

Ni-Ni bond length = 2.457 Å  
Zr-Zr bond length = 3.170 Å

$$5 \times 2.457 = 12.28 \text{ Å}$$
$$4 \times 3.170 = 12.68 \text{ Å}$$

Parameter ratio  $a_{\text{Ni}}/a_{\text{Zr}} = 1.070$

Minimize the stress in xy plane!

Ni(111): 9 x 9    22.113 Å  
Zr(0001): 7 x 7    22.190 Å



Lattice aspect  
ratio ~ 1



Ni(111): 5 x 5    12.28 Å  
Zr(0001): 4 x 4    12.68 Å

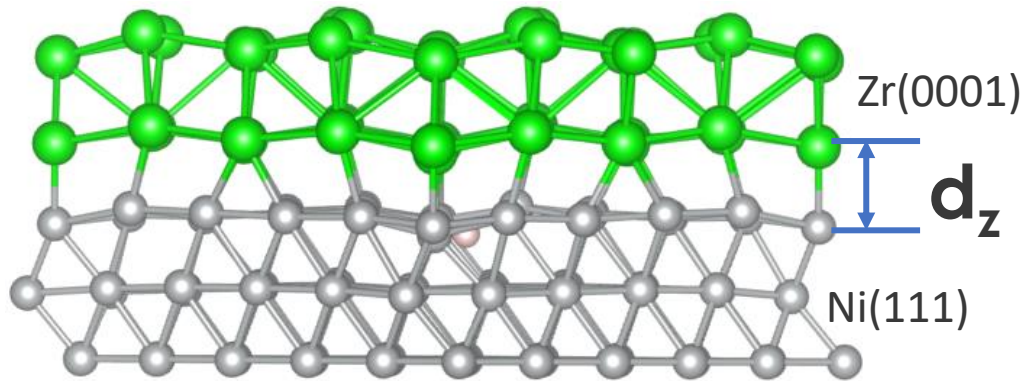
Parameter ratio ~ 0.996

Parameter ratio ~ 1.070

# Zr/Ni Interface Models

## Ni(111)/Zr(0001) Interfaces and Model Optimization

### Five-Layer Ni()/Zr Interface Model

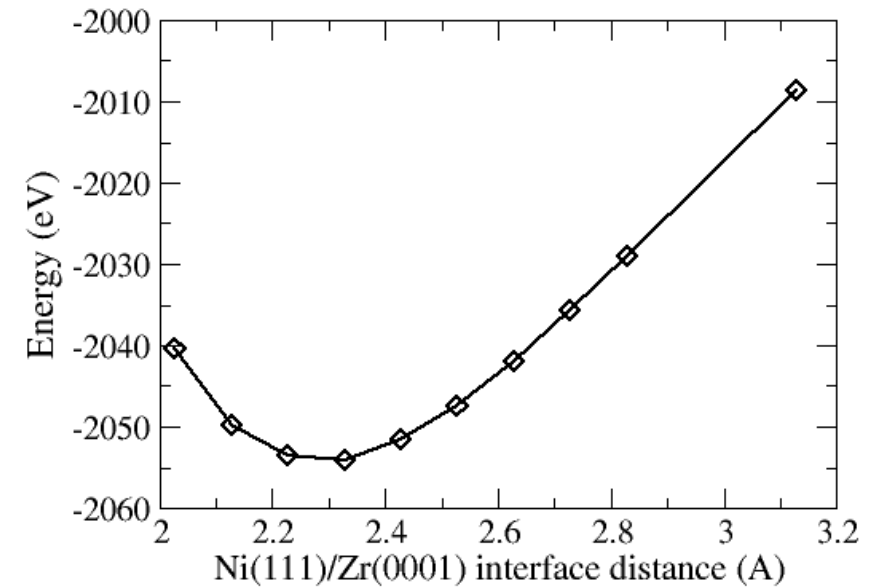


$$5 \times 2.457 = 12.28 \text{ \AA}$$

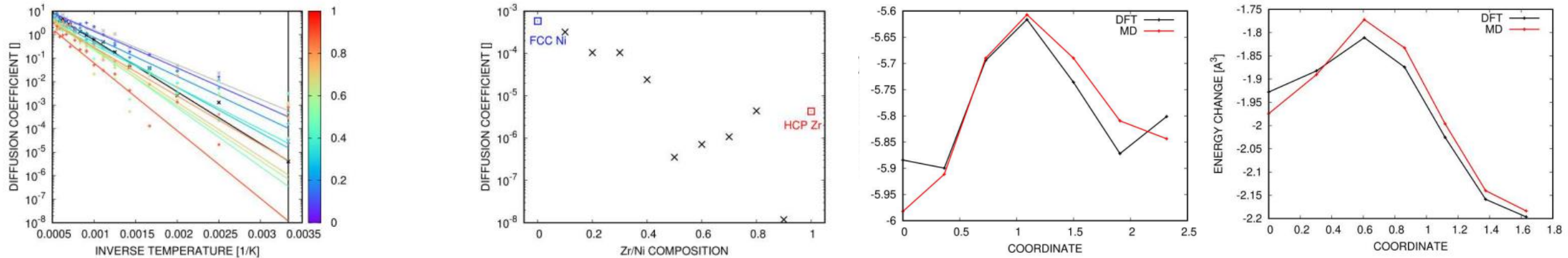
$$4 \times 3.170 = 12.68 \text{ \AA}$$

$$\text{New ratio } a_{\text{Ni}}^*/a_{\text{Zr}}^* = 1.032$$

### Interface Optimization



$$d_{opt} = 2.32 \text{ \AA}$$

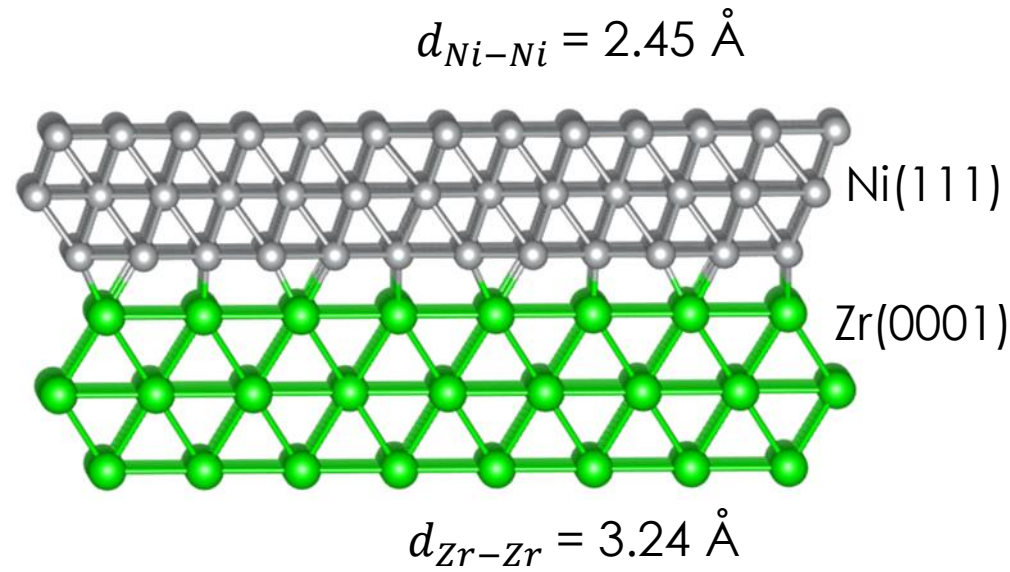


*Jones et al., SAND2019-15034R*

- Ni atoms diffuse crossing the interface more easily and deeply into Zr side than Zr atoms into Ni side.
- The activation energies of Ni and Zr are 1.25 and 1.28 eV for Ni(100)//Zr(0001) interface, 1.33 and 1.42 eV for Ni(110)//Zr(0001) interface at the temperature range of 950–1100 K.
- The interdiffusion for case of Ni(110)//Zr(0001) interface is easier than that of Ni(100)//Zr(0001) interface.

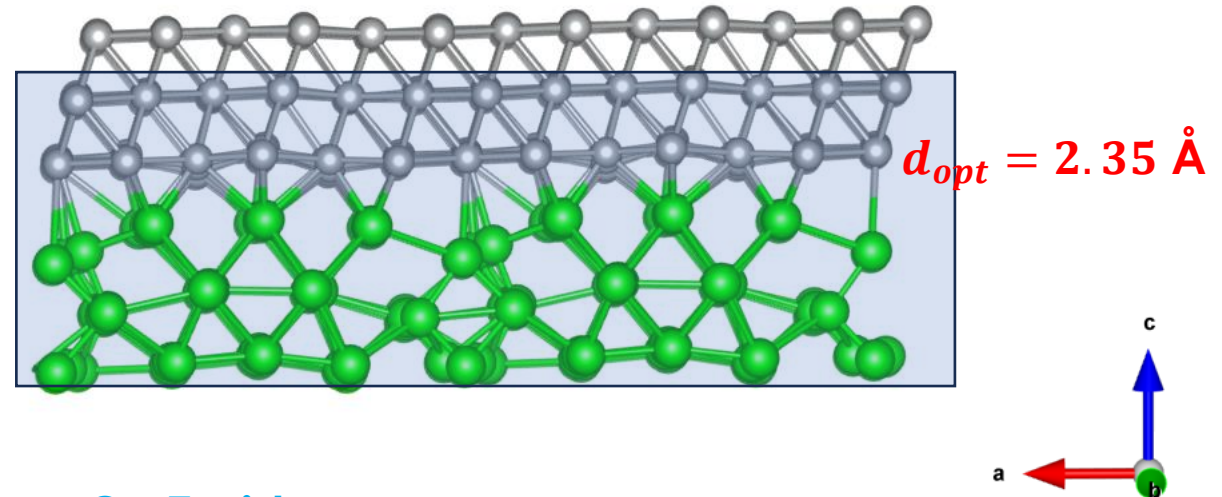
# Ni(111)/Zr(0001) Interface Model

## Six Layers Ni(111)Zr(0001) Interface Model



### On Ni side:

- Bond distance varies  $d_{Ni-Ni} = 2.5-2.65 \text{ \AA}$
- More displacement along a-axis
- Minimal displacement along c-axis



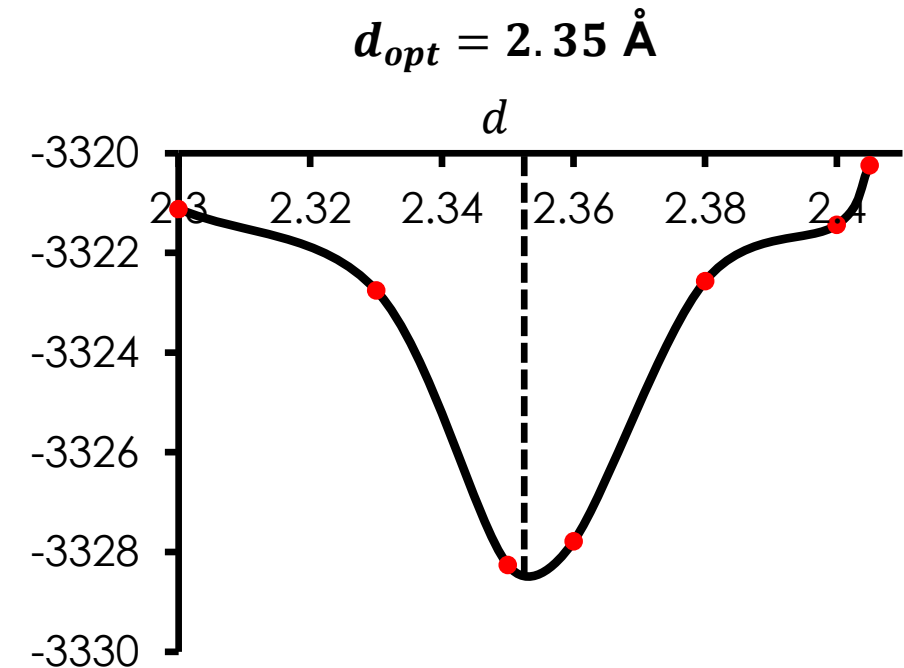
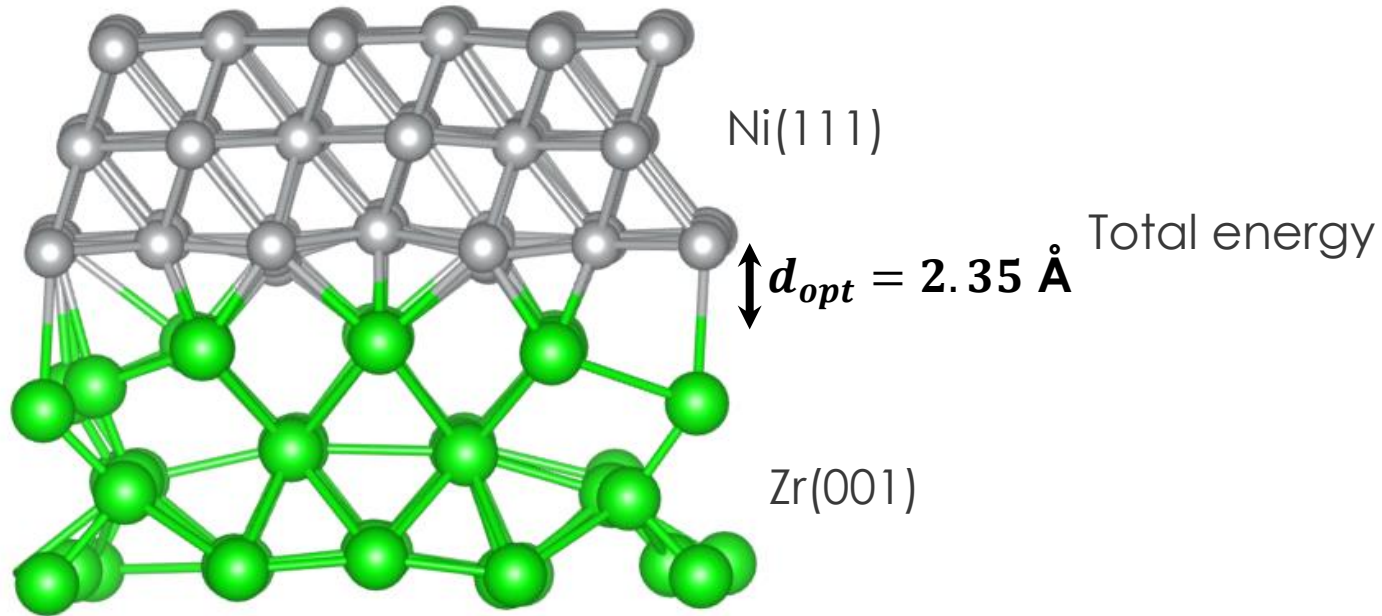
### On Zr side:

- More displacement along a-axis
- Boundary atoms are more displaced
- Displacement of some atoms by more than an Å



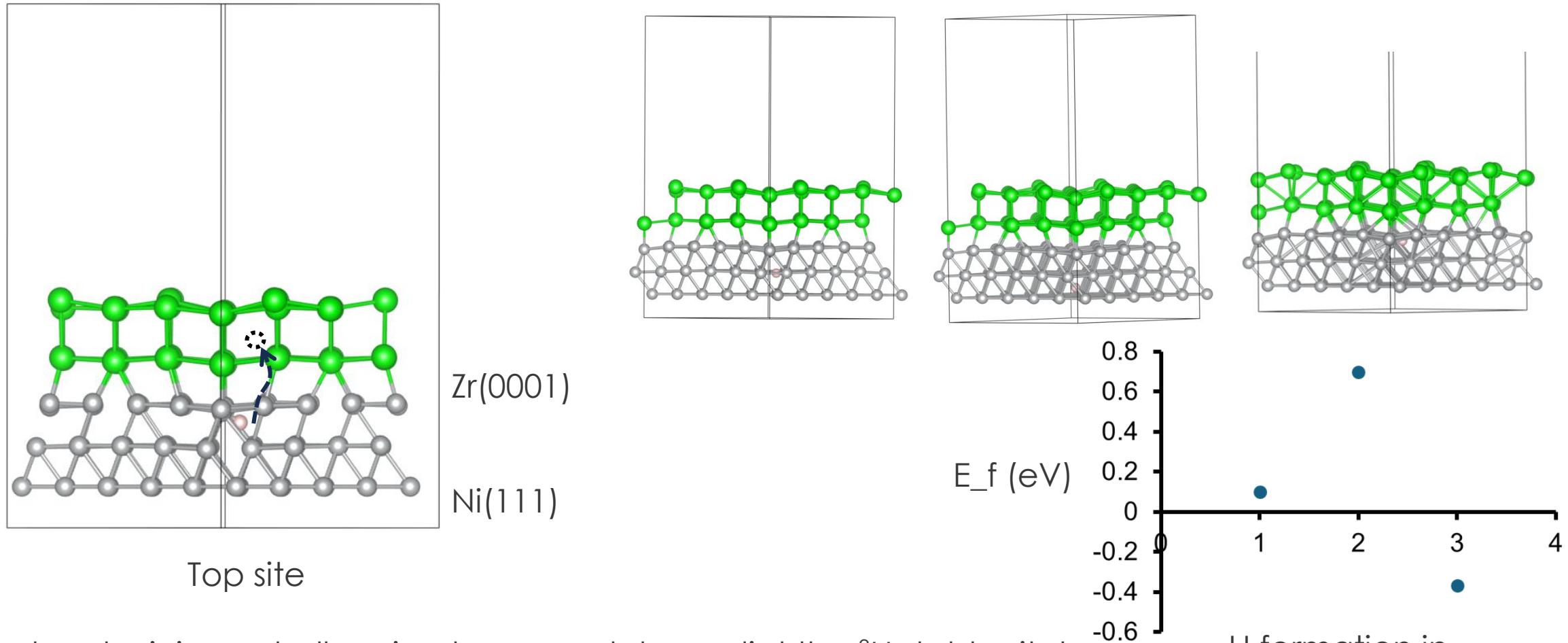
# Ni(111)/Zr(0001) Interface Model

## Six Layers Ni(111)Zr(0001) Model Optimization



# Ni(111)/Zr(0001) Interface Model

## Tritium Formation at Ni (111)/Zr(0001) - Initial State Optimization

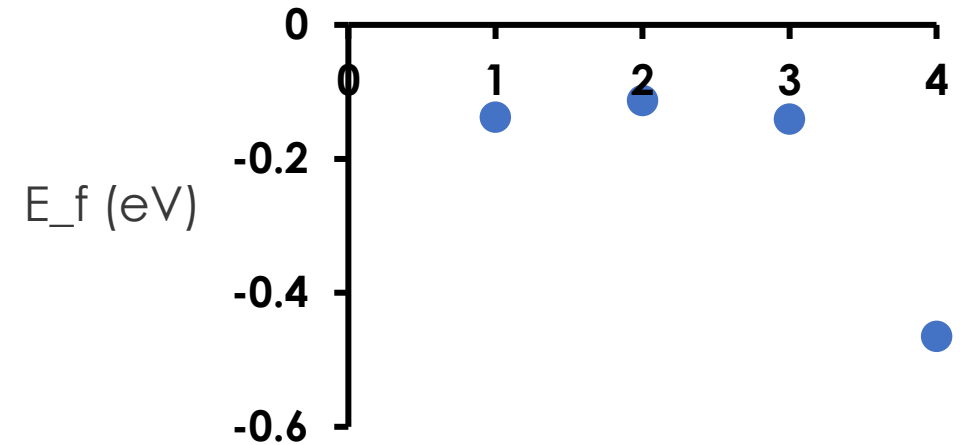
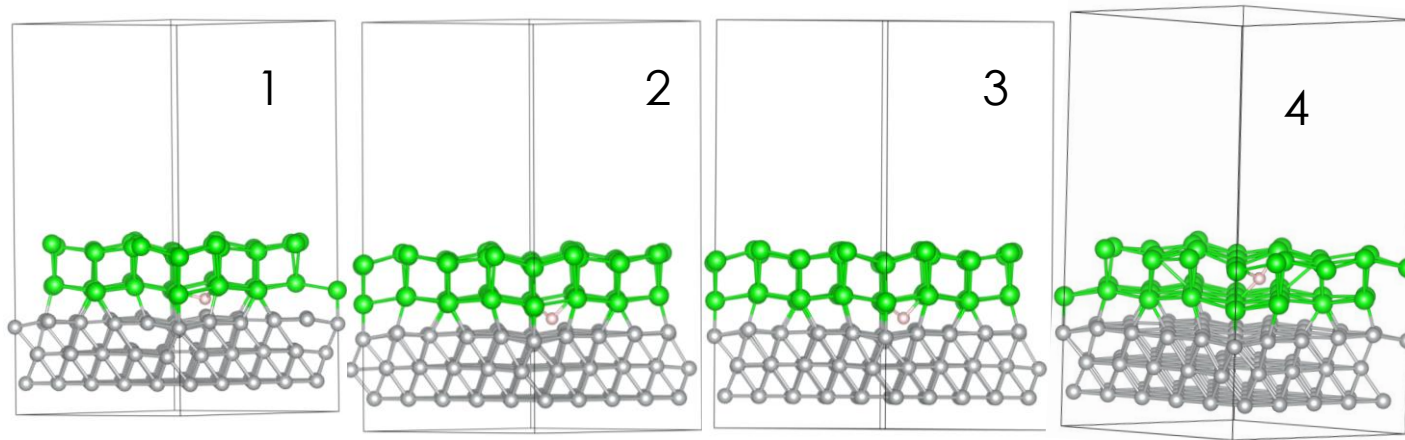


Many local minima, challenging to accurately predict the  $^3\text{H}$  stable site!

H formation in  
Ni site = -0.36 eV

# Ni(111)/Zr(0001) Interface Model

## Tritium Formation at Ni (111)/Zr(0001) - Final State Optimization



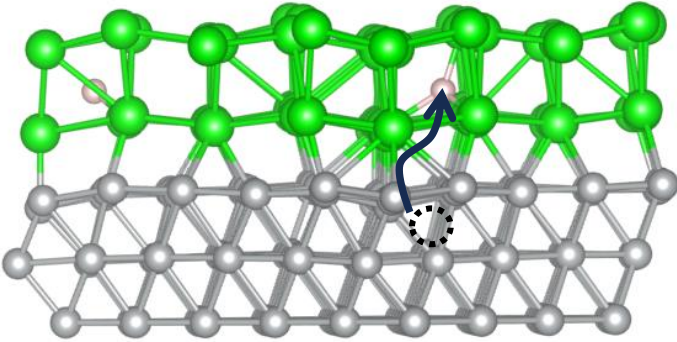
H formation in  
Zr site = -0.46 eV

Many local minima, challenging to accurately predict the  $^3\text{H}$  stable site!

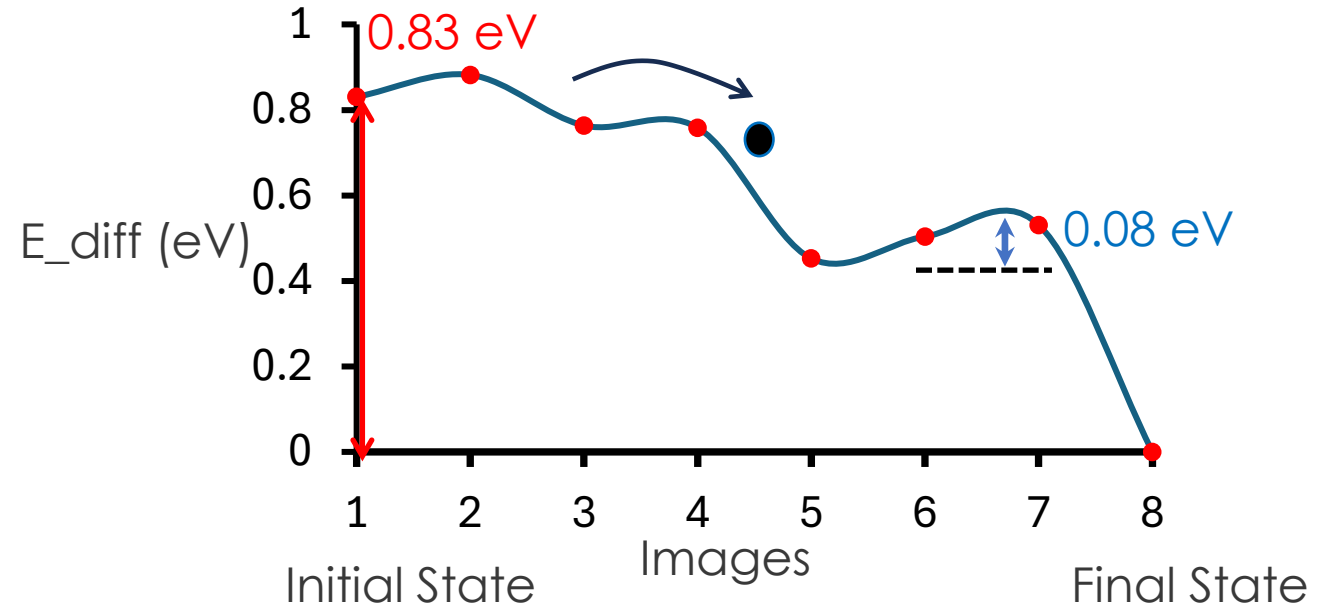
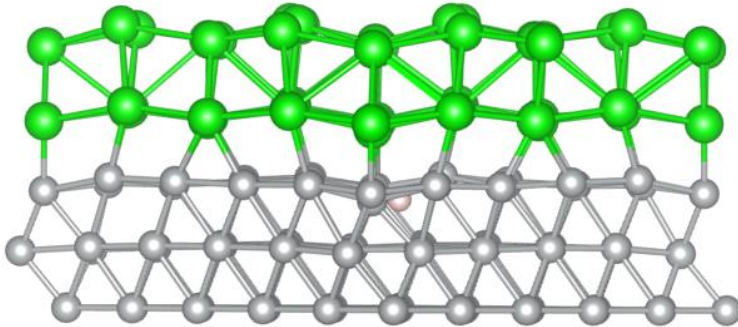
# Ni(111)/Zr(0001) Interface Model

## Tritium Diffusion Across Ni(111)/Zr(0001) Interface

Final State



Initial State

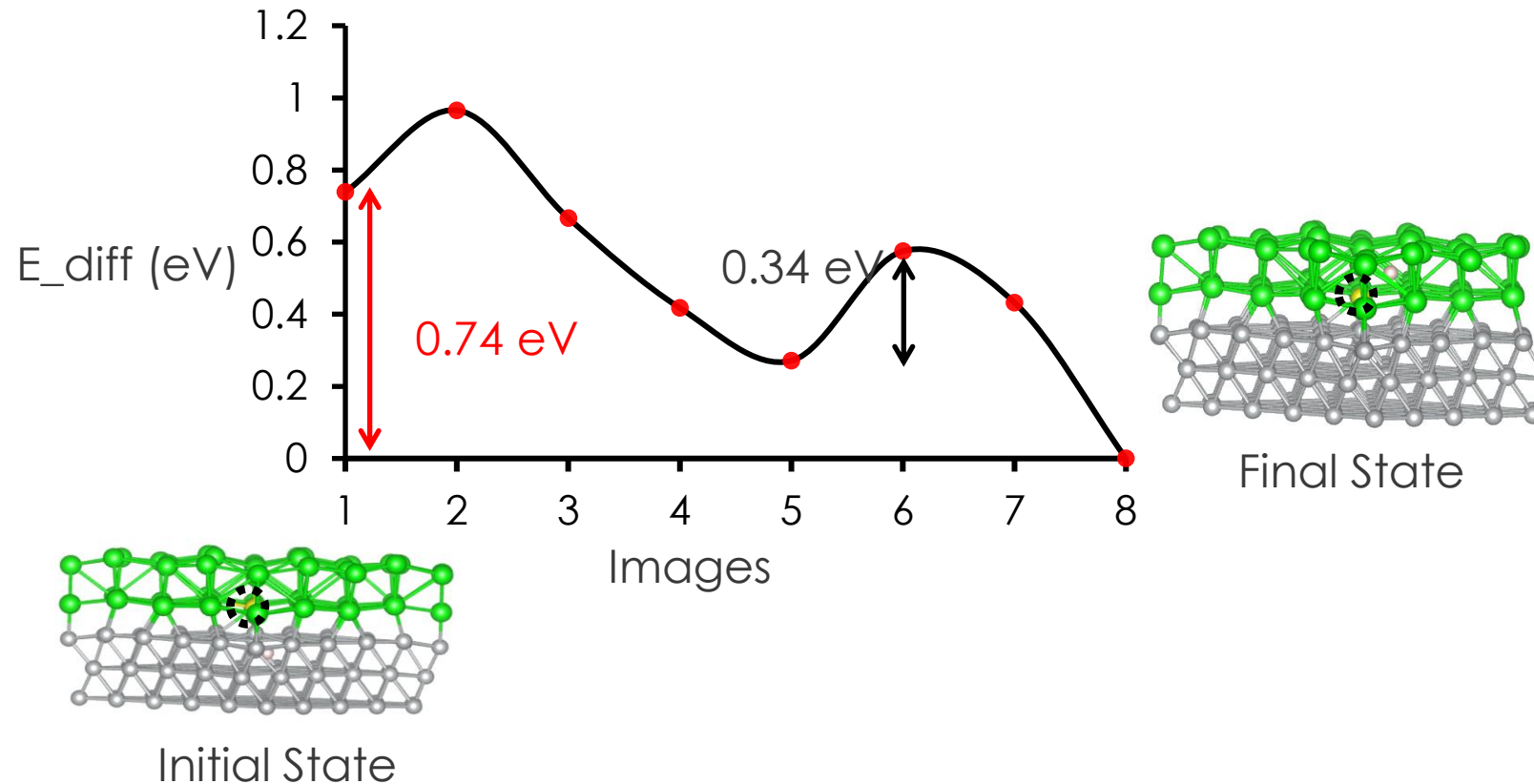


Zr (0001) site for  $^3\text{H}$  is 0.83 eV lower than Ni (111) site.



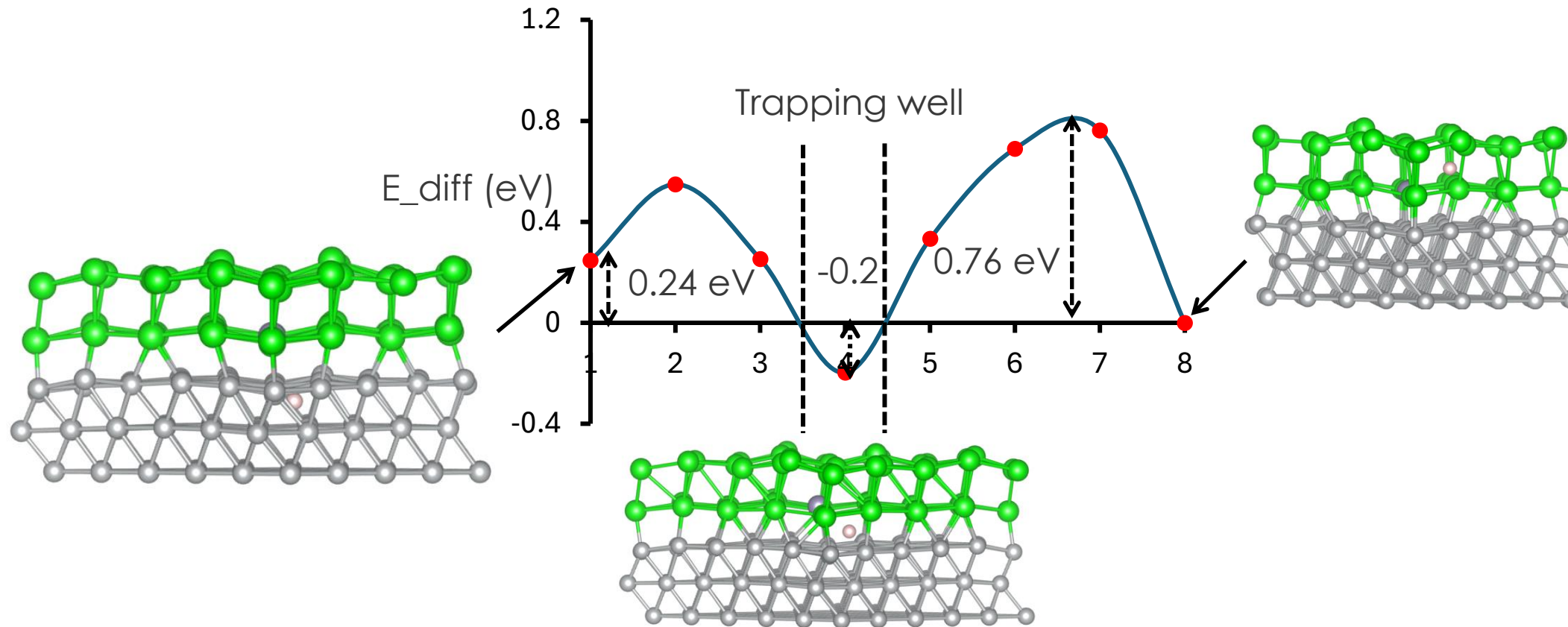
# Ni(111)/Zr(0001) Interface Model

## Tritium Diffusion Across Ni(111)/Zr(0001) Interface: With One Zr Vacancy



# Ni(111)/Zr(0001) Interface Model

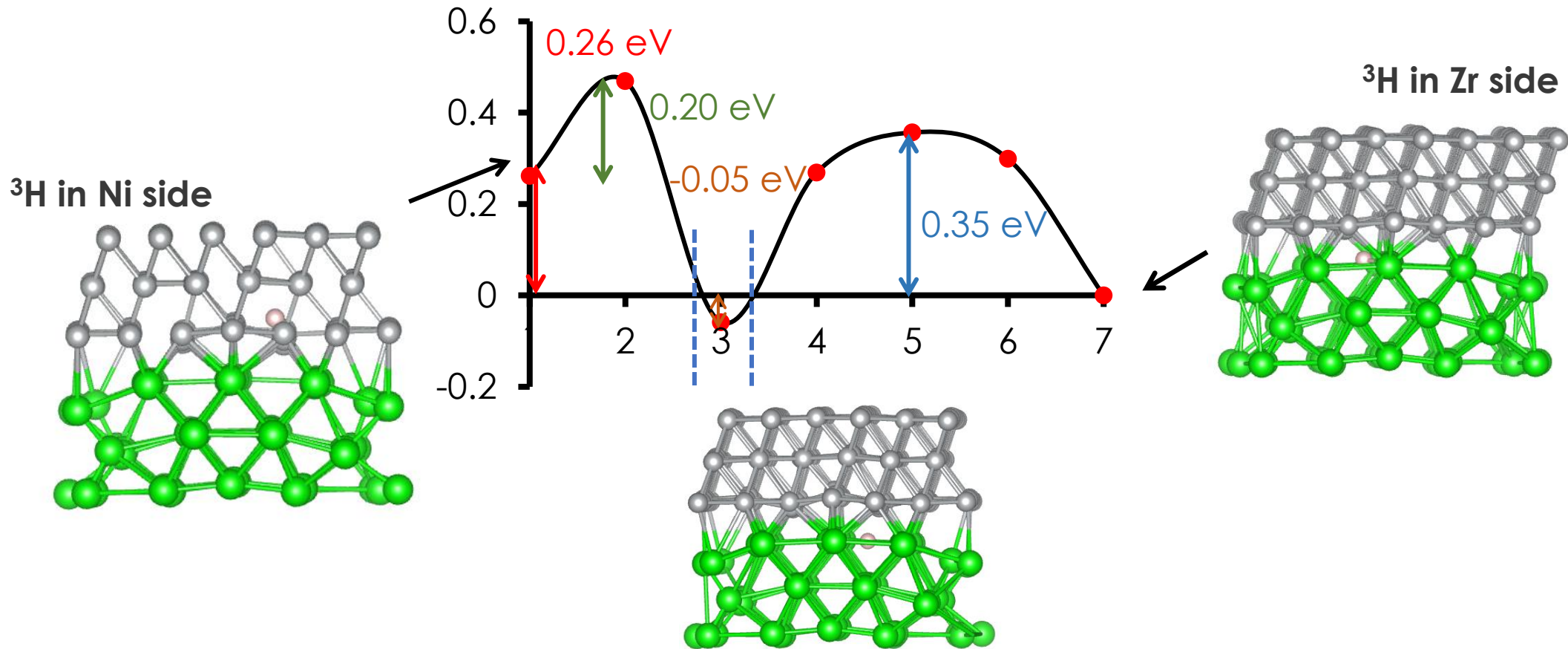
## Tritium Diffusion Across Ni(111)/Zr(0001) Interface: With a Sn Impurity



With a Sn impurity, small  $^3\text{H}$  trapping well was found at the interface.

# Ni(111)/Zr(0001) Interface Model

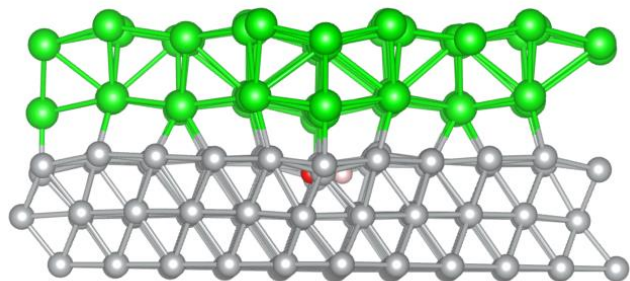
## Diffusion of H in Ni(111)/Zr(001) Six Layers



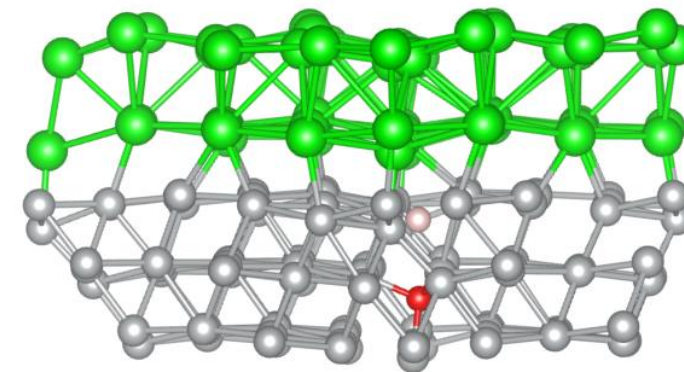
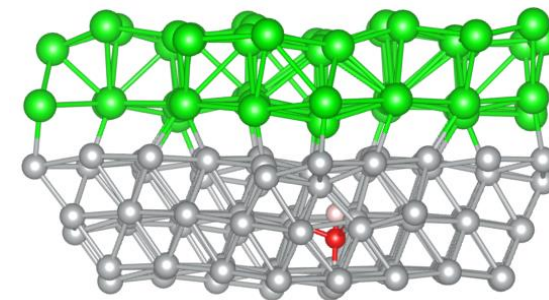
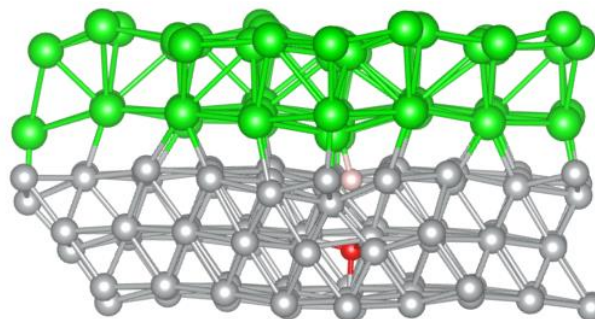
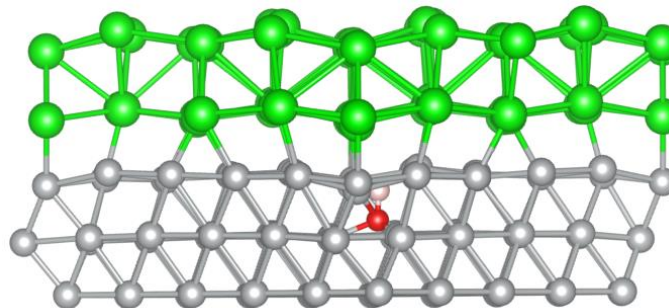
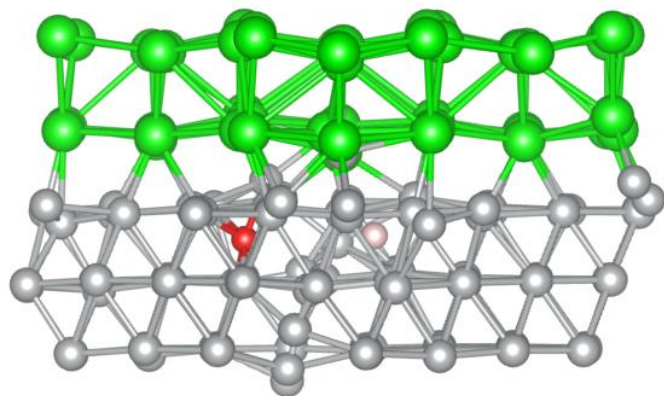
$^3\text{H}$  is more stable by 0.26 eV in Zr region than in Ni!

# Ni(111)/Zr(0001) Interface Model

## Introducing Ni(OH) at the Interface



Optimization



**<sup>3</sup>H Formation** -4.87 eV

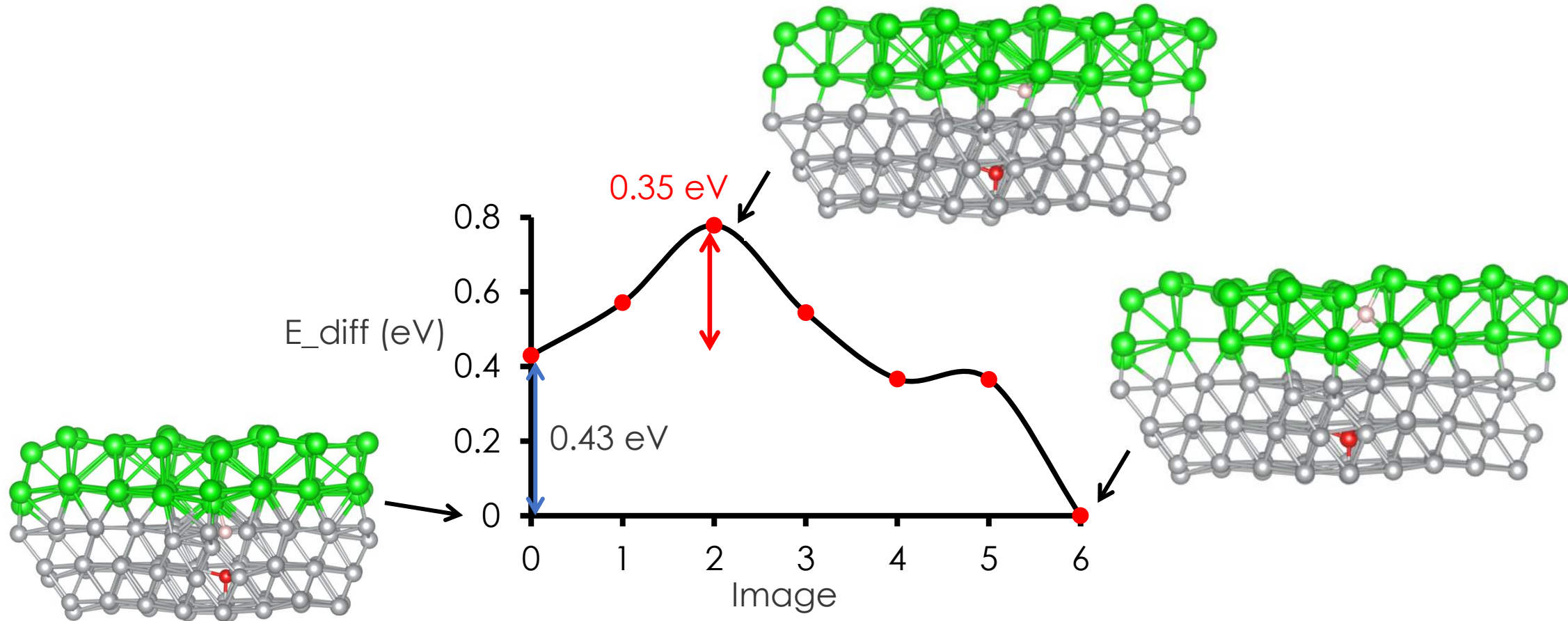
-6.64 eV

-6.59 eV



# Ni(111)/Zr(0001) Interface Model

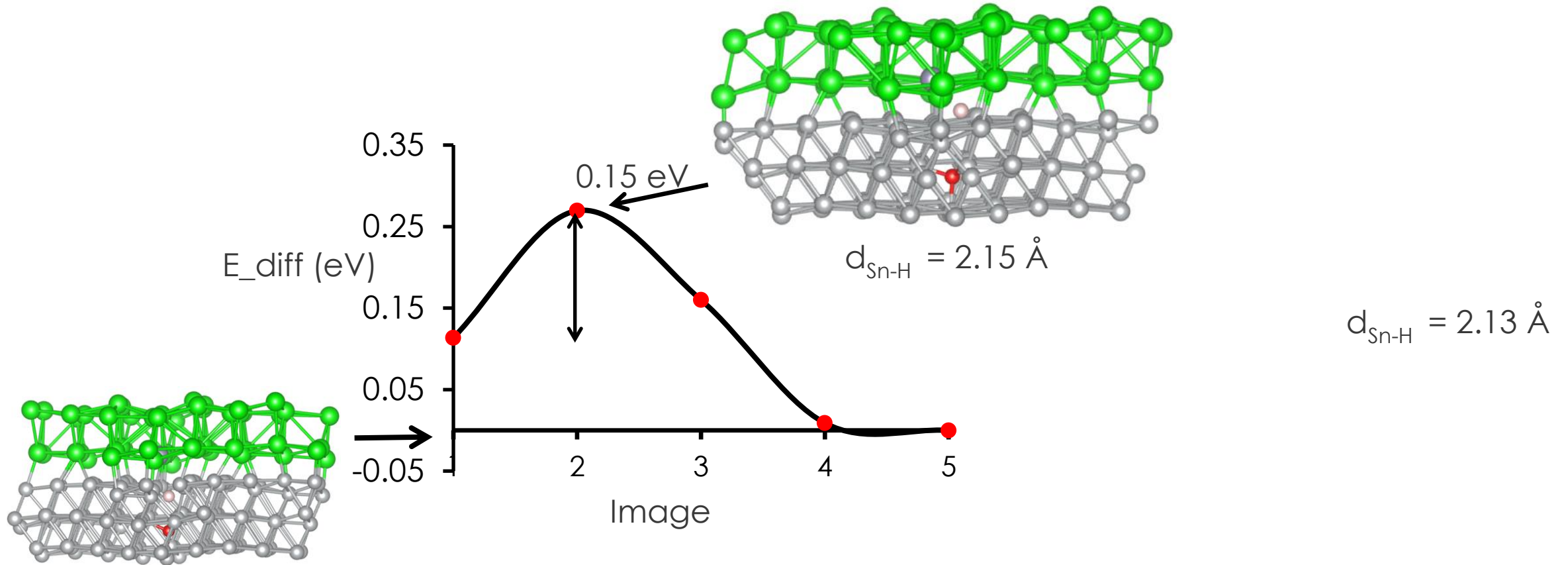
## Introducing Ni(OH) at the Interface: Diffusion of $^3\text{H}$



Initial energy barrier height for tritium diffusion in presence of O in Ni (111) was found to be 0.35 eV, almost 0.25 eV higher than with no O impurity but  $^3\text{H}$  is more stable by 0.43 eV in Zr region.

# Ni(111)/Zr(0001) Interface Model

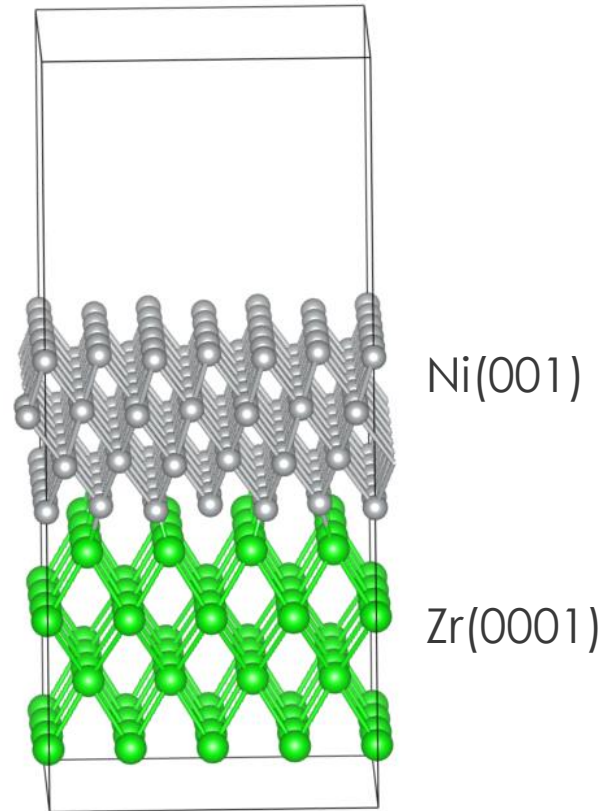
## Introducing Sn and Ni(OH) at the Interface: Diffusion of $^3\text{H}$



The  $^3\text{H}$  stability in Zr region is reduced by 0.2 eV with Sn and Ni(OH) as compared to with Ni(OH) only.

# Ni(111)/Zr(0001) Interface Model

## Ni(001)/Zr(0001) Interface Model



This model provides a more realistic picture for the  $^3\text{H}$  diffusion at Ni/Zr interface. However, it requires intensive computational resources for the diffusion property calculations.

# Ni/Zr Alloy Interface Development

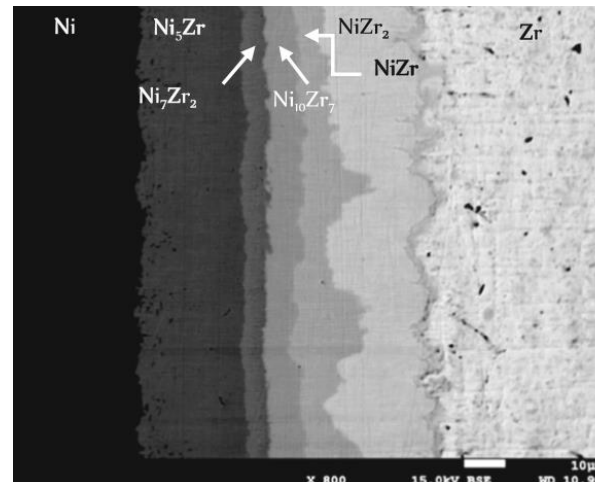
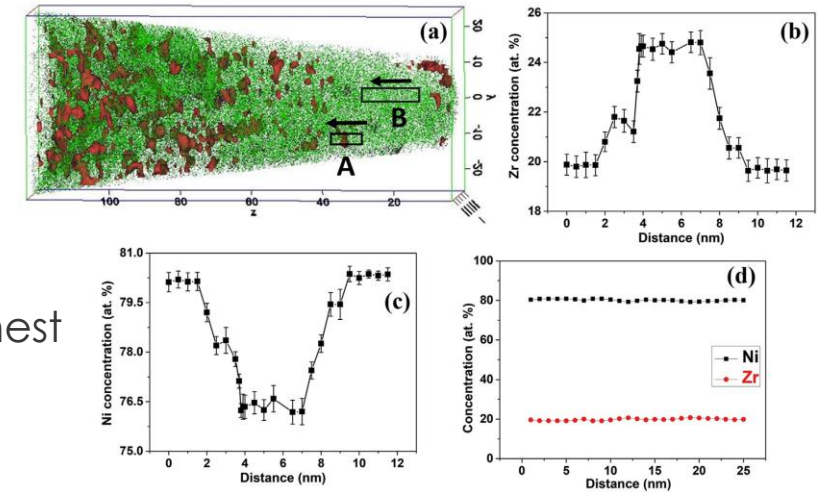
## Simple Toy Model for Ni(111)/Zr(0001) Alloy at the Interface

Ni-Ni bond length = 2.457 Å  
Zr-Zr bond length = 3.170 Å

Five intermetallic compounds,  $\text{Ni}_5\text{Zr}$ ,  $\text{Ni}_7\text{Zr}_2$ ,  $\text{Ni}_{10}\text{Zr}_7$ ,  $\text{NiZr}$  and  $\text{NiZr}_2$  are found to grow in the interdiffusion zone.

Activation energy is found to be lowest for  $\text{Ni}_{10}\text{Zr}_7$  ( $178 \pm 8$  kJ/mol) and highest for  $\text{NiZr}$  ( $323 \pm 6$  kJ/mol).

In amorphous  $\text{NiZr}$  smaller atoms are the predominant diffusing species.



**Machine learning assisted computational model for complex interfacial Ni/Zr alloy with varying elemental composition.**

TEM studies for microstructure evolution of the as-deposited film (EDXS spectrum showing the Ni and Zr elements).

Back-scattered electron (BSE) image of the interdiffusion zone.

*Journal of Alloys and Compounds* 844 (2020) 156078  
*Journal of Phase Equilibria and Diffusion* 36, 4, 2015



# Summary and Conclusions

- We created Ni/Zr interface models with different Ni and Zr surfaces.
- We optimized the interface models for Ni(111)/Zr(0001) with total of five and six layers and studied the tritium diffusion pathways.
- Ni-Zr bond length was optimized and was found to be 2.32 Å for five layer and 2.35 Å for six layers interface systems.
- In Ni/Zr interface model,  $^3\text{H}$  was found to be relatively more stable by 0.83 eV in Zr (0001) region than Ni (111) region.
- With a substitutional Sn in in Zr-site, the relative stability was found to be 0.24 eV higher in Zr(0001) region.
- $^3\text{H}$  was found to trap by an energy 0.2 eV at the interface region with Sn impurity.
- Initial energy barrier height for tritium diffusion in presence of O in Ni (111) was found to be 0.35 eV, almost 0.25 eV higher than with no O impurity but  $^3\text{H}$  was more stable by 0.43 eV in Zr region.
- The  $^3\text{H}$  stability in Zr region was reduced by 0.2 eV with Sn and Ni(OH) as compared to with Ni(OH) only.

# Thank You!

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