

Martensitic transformations and shear-band interactions in austenitic stainless steel: effects of hydrogen

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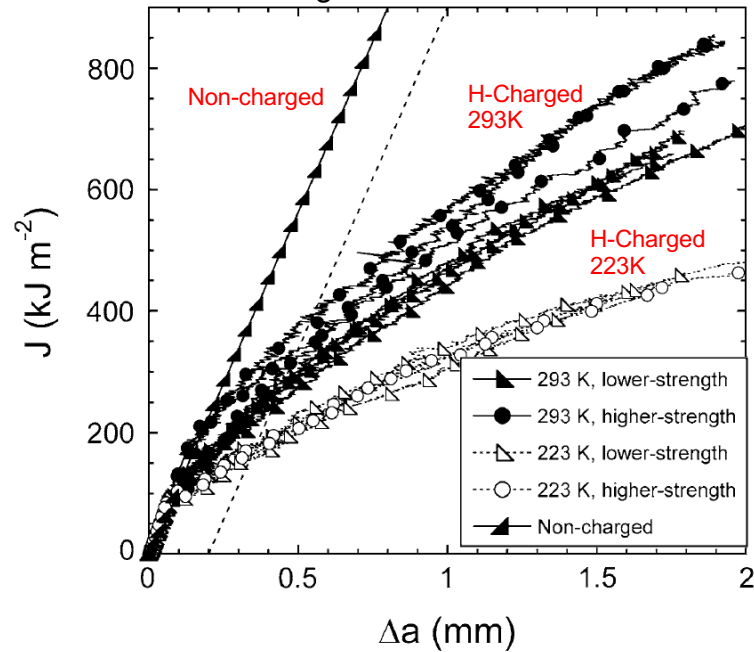
SAND2023 - tbd



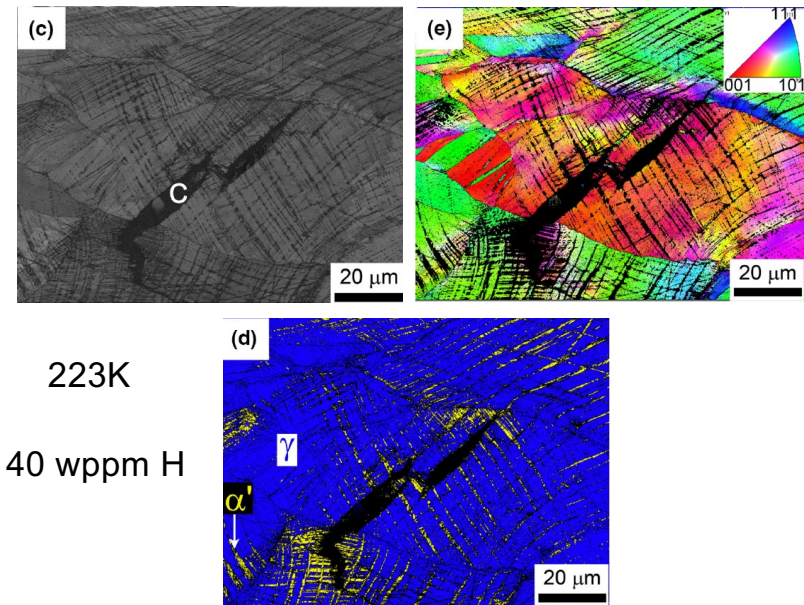
Hydrogen degrades fracture toughness of Austenitic Stainless Steel

Crack Growth Resistance (J-R) Curve

Forged 304L Stainless Steel



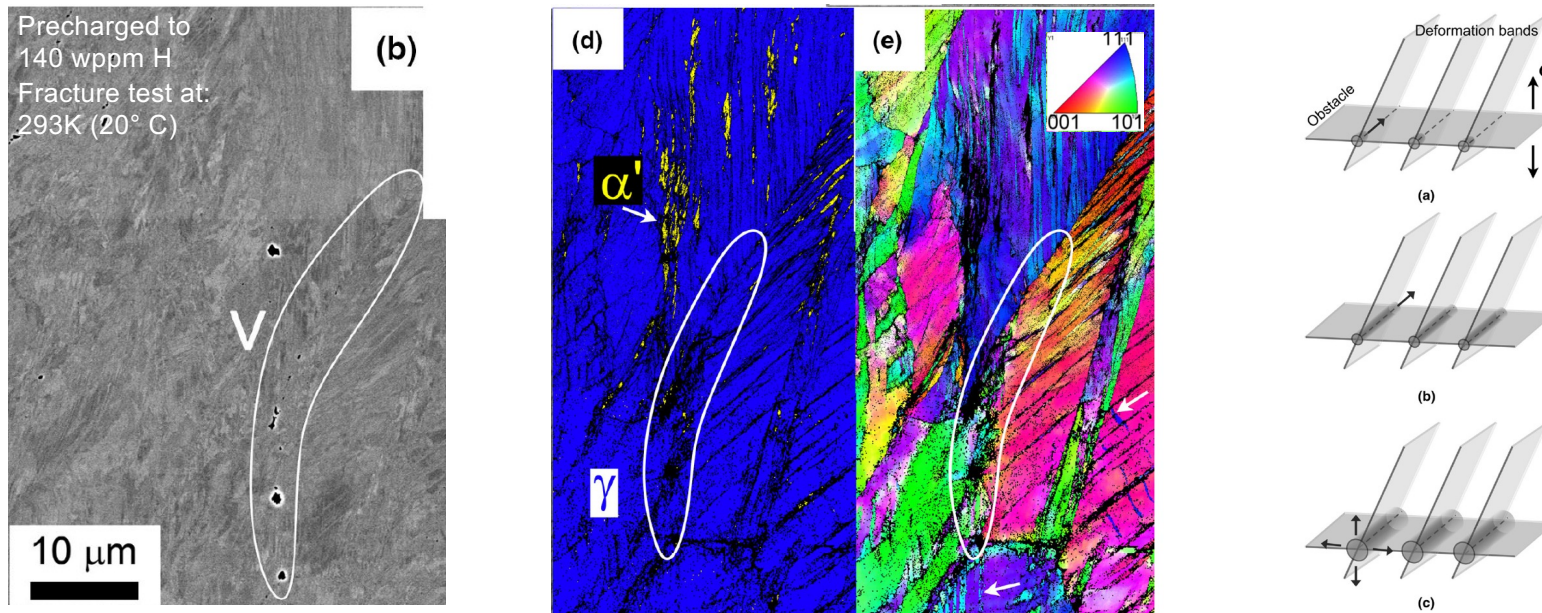
Microvoid initiation at intersecting deformation-bands



H. Jackson, C. San Marchi, D. Balch, B. Somerday, J. Michael, Metallurgical and Materials Transactions A, 2016

Microvoid nucleation at planar deformation band intersections with grain boundaries

Fracture processes in H-Charged 304L associated with void nucleation

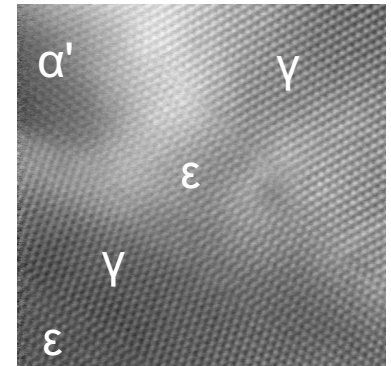
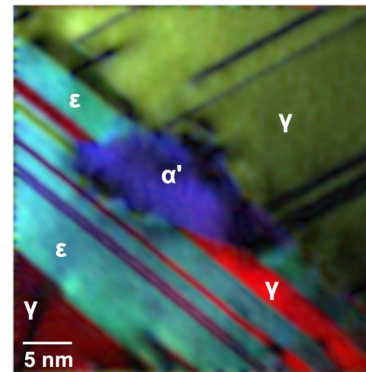
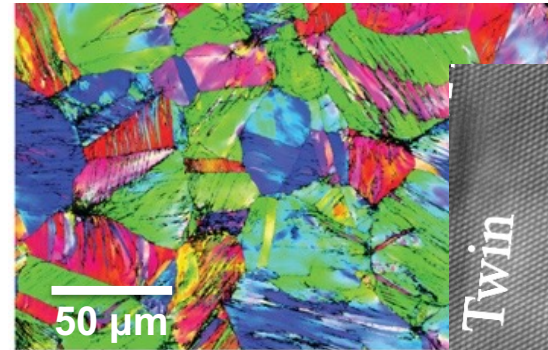


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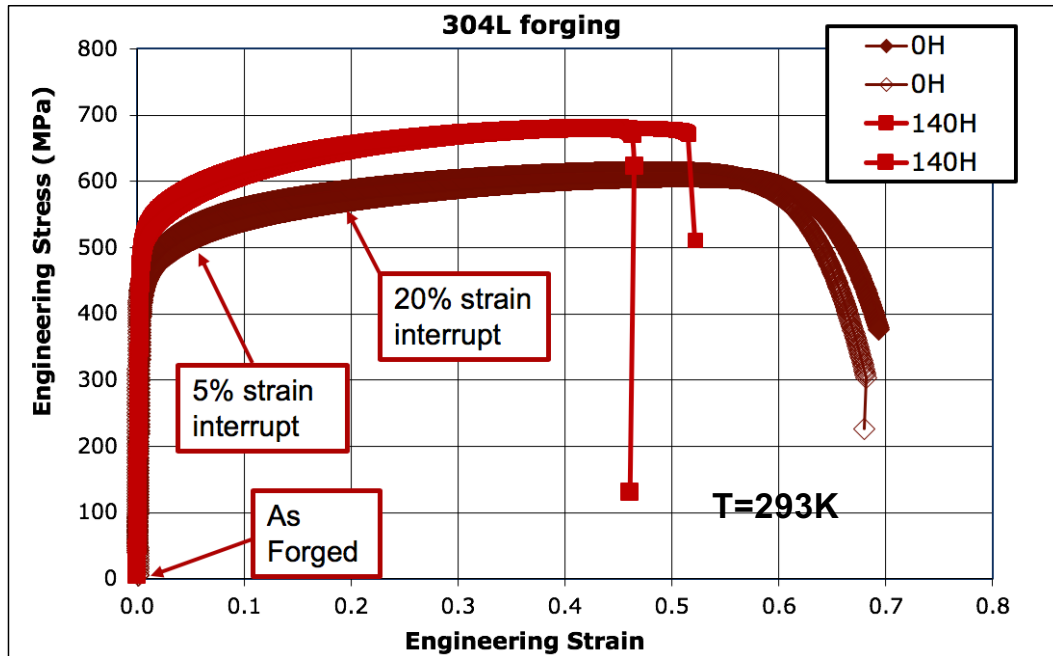
What are the deformation bands and how do they depend on hydrogen?

Focus for this talk:

- **Structure of the planar deformation bands**
 - Influence of hydrogen on formation of ϵ -martensite
- **Intersection of shear-bands**
 - nucleation of α' martensite
 - twinning at intersecting ϵ -laths



Material: Forged 304L Austenitic Stainless Steel

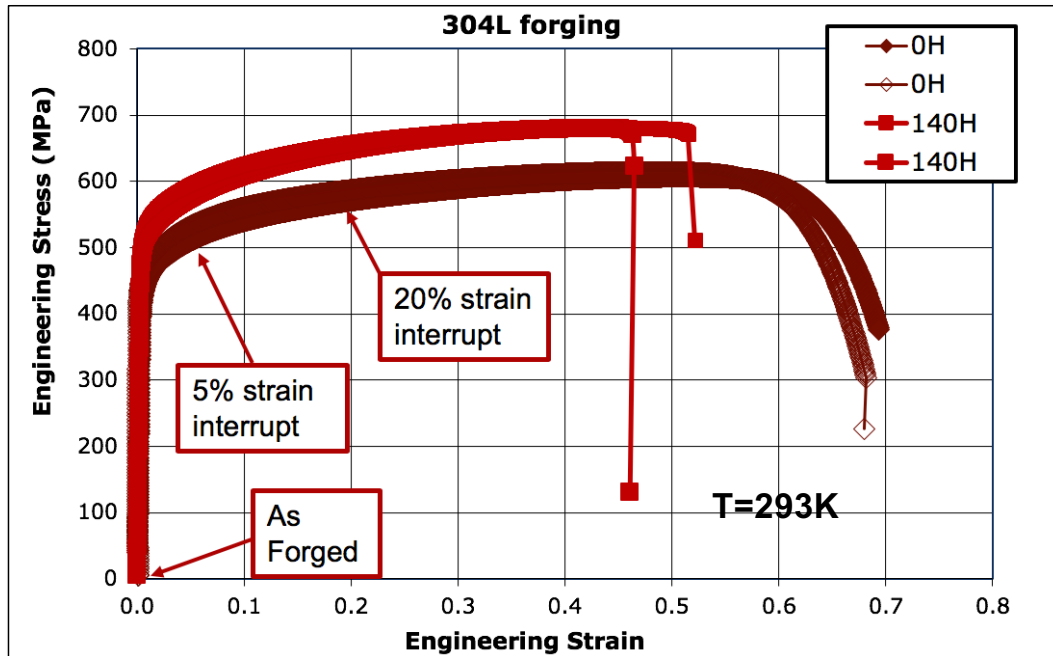


- Initial state: Forged, strain-hardened condition
- H-Charged material:
 - Increased YS and UTS
 - Reduced ductility

Hydrogen Pre-charging: 140 wppm Same starting forged material
 2 wks, $P_{H_2}=138$ MPa, $T=300^\circ\text{C}$ as in Jackson's 2016 study

Fe	Cr	Ni	Mn	Si	C	N	P	S
Bal	19.64	10.6	1.62	0.65	0.028	0.04	0.02	0.0042

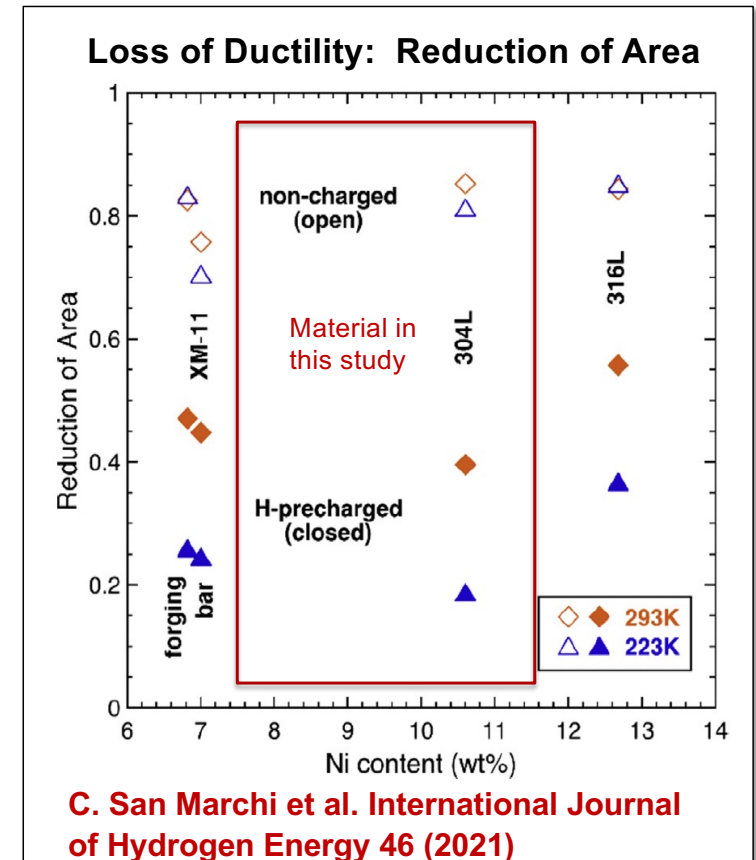
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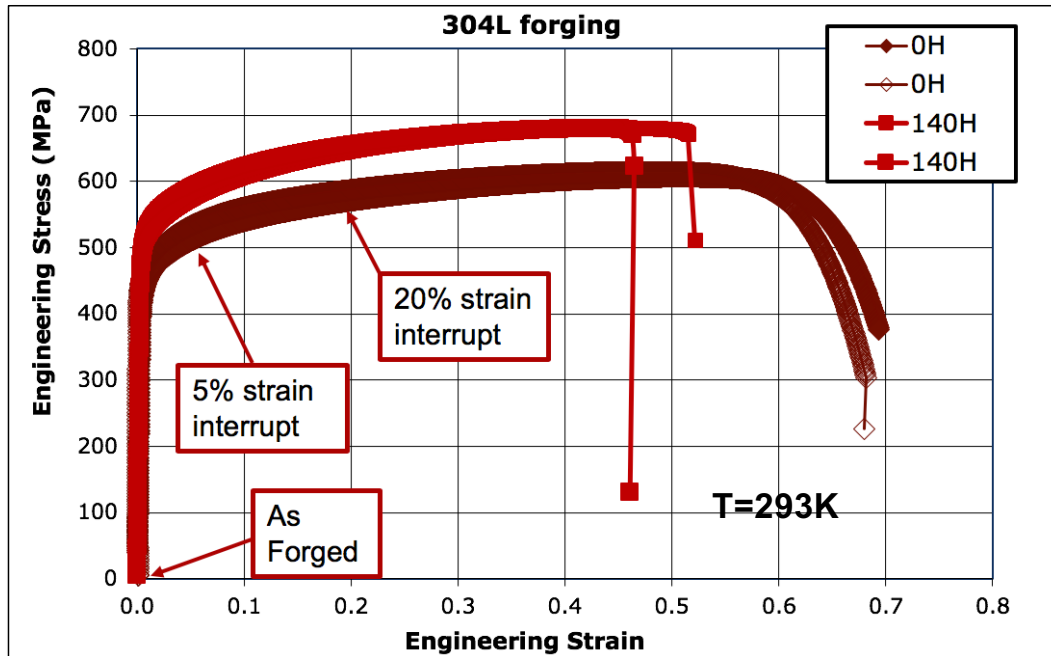
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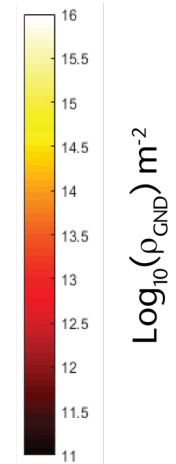
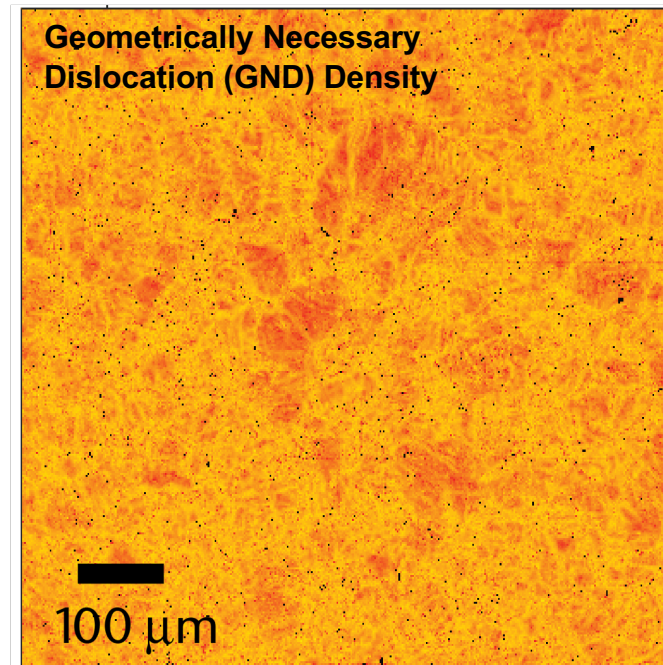
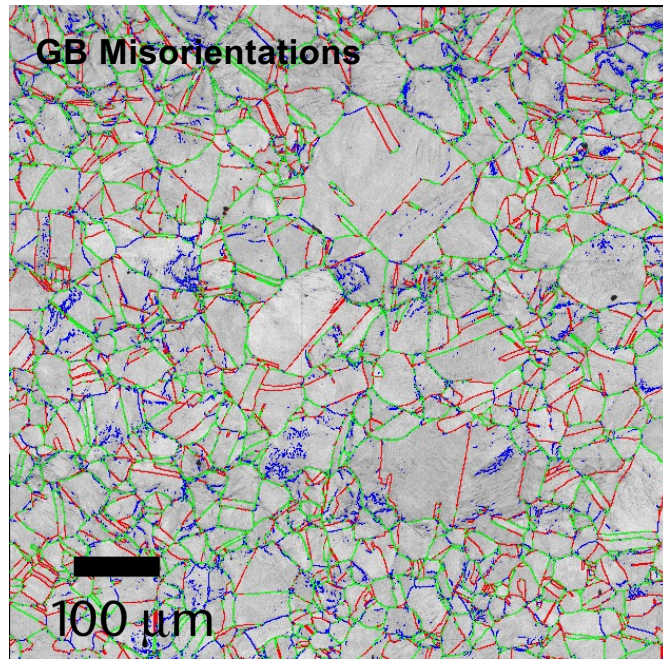
- Microstructure analysis from interrupted tensile tests

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Initial As-forged microstructure

EBSD Measurements



Misorientation

5-20° ————

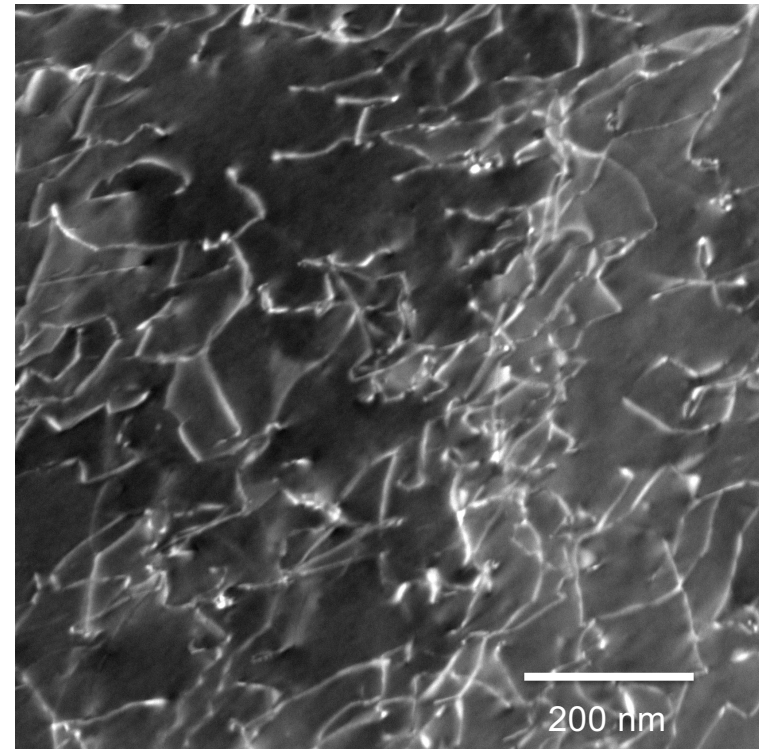
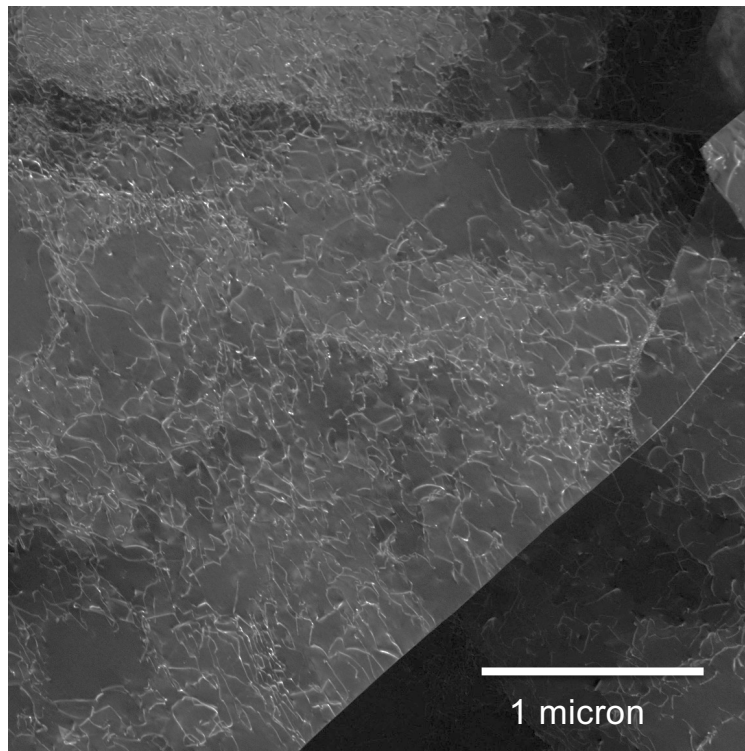
20°-55° ————

55°-60° ————

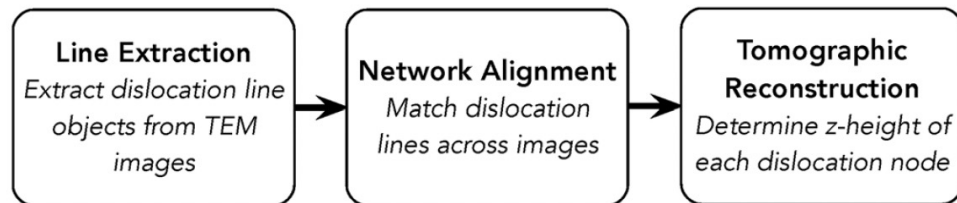
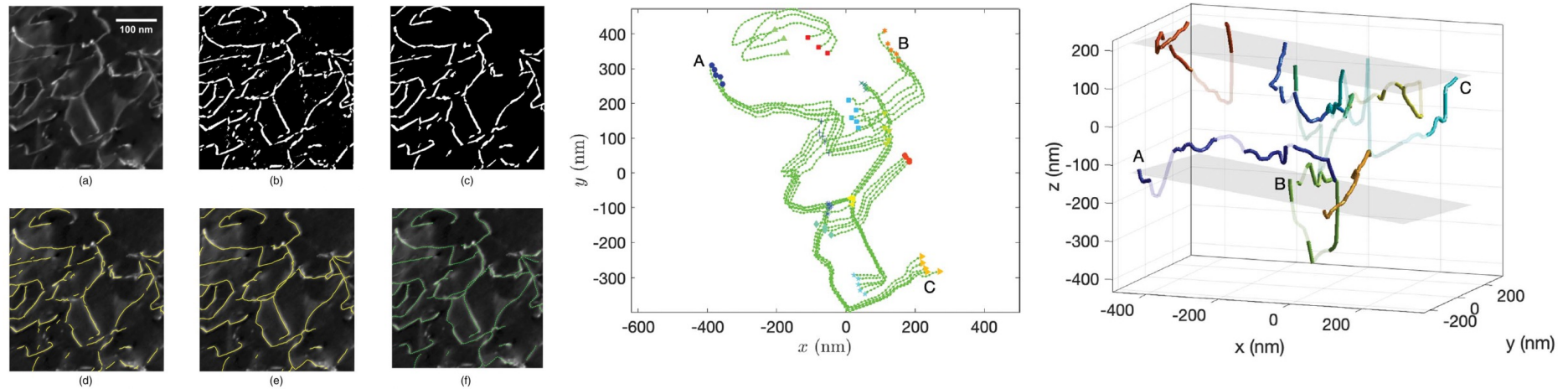
- Some pre-existing twins within the microstructure
- Dense distribution of geometrically necessary dislocations (GNDs)

As-forged microstructure: dense dislocation network

Diffraction Contrast Scanning Transmission Electron Microscopy
(DC-STEM)



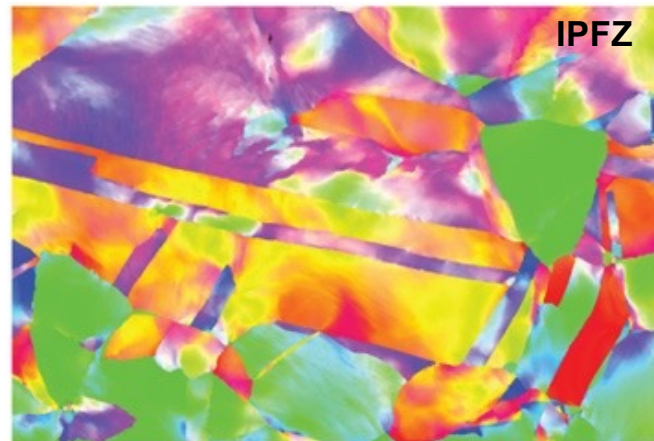
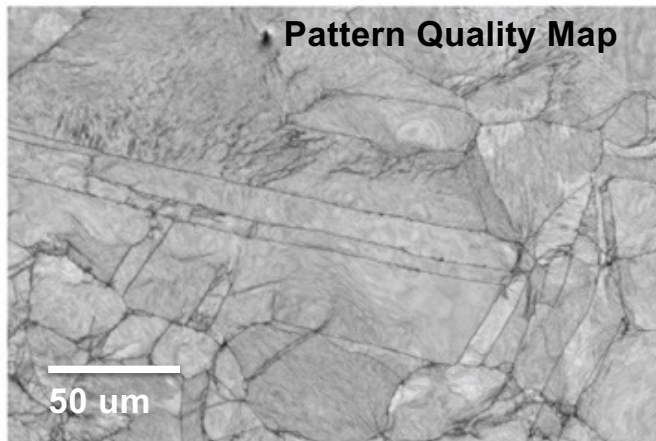
Object-Based Dislocation Tomographic Reconstruction



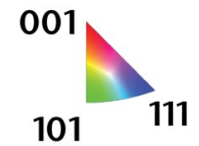
Influence of H on deformation

EBSD - 20% Strain

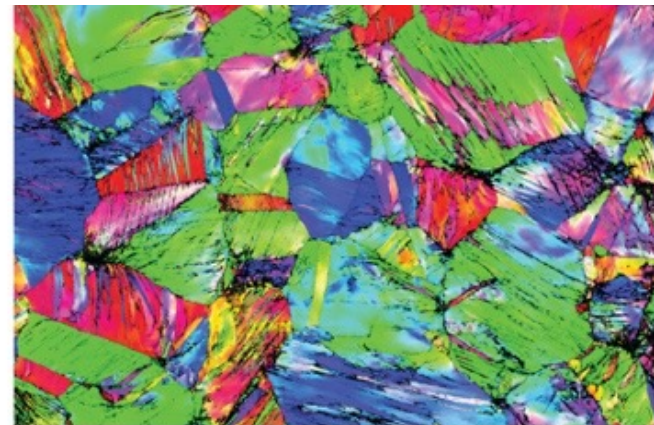
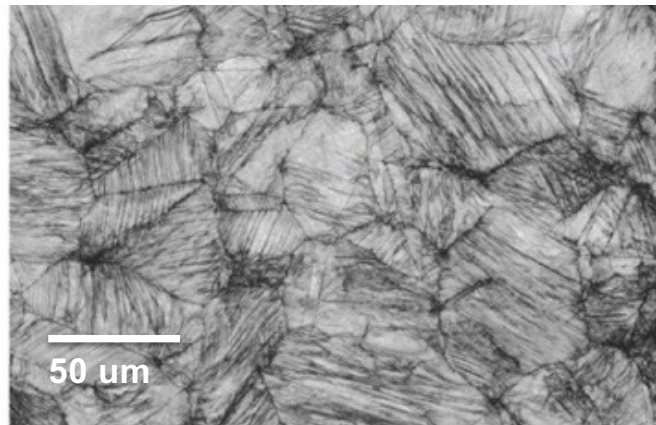
Non-
Charged



T = 293K



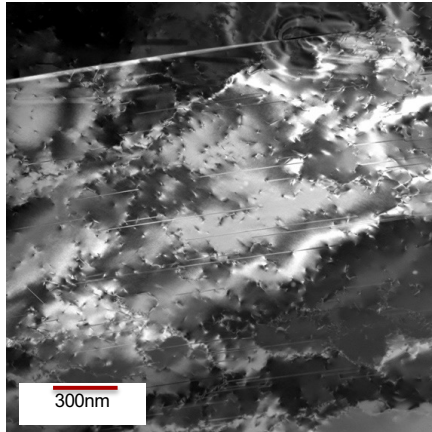
Hydrogen
Charged



Hydrogen
increases
density of
deformation
bands

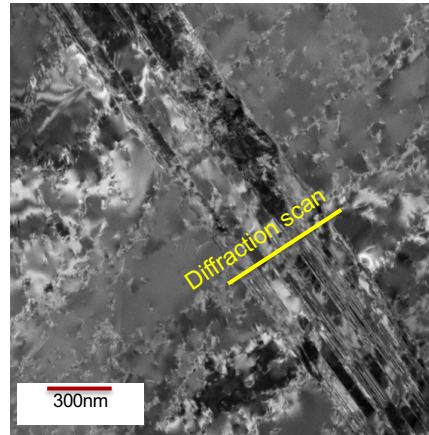
STEM: Insight to Development of Shear Bands

As-forged and H-charged



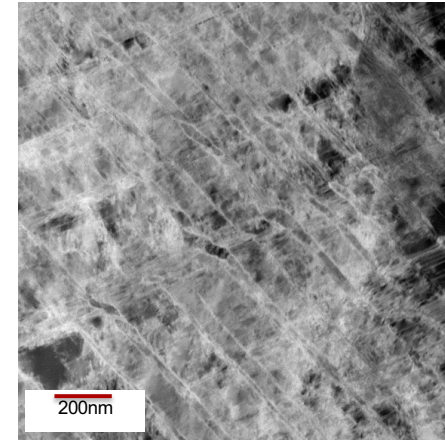
Dislocation cells and extended stacking faults

5% strain (140 ppm H)



Parallel bands of deformation twins and ϵ -martensite (no α' -martensite observed)

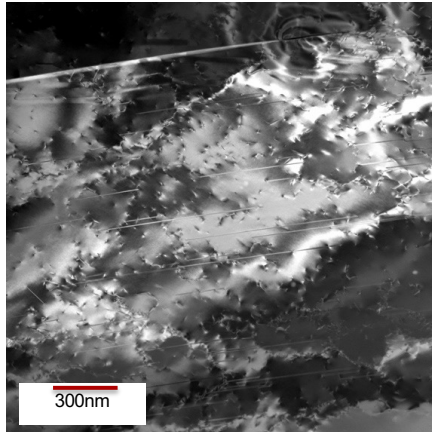
20% strain(140 ppm H)



Intersecting shear bands (twins, ϵ -martensite)
 α' – martensite at intersections

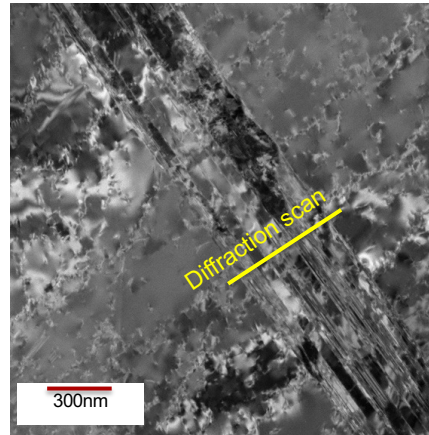
STEM: Insight to Development of Shear Bands

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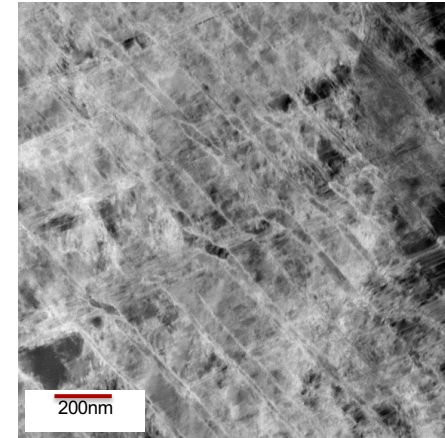
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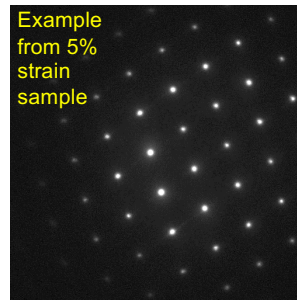
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Intersecting shear bands (twins, ϵ -martensite)
 α' – martensite at intersections

*Scanning
diffraction to
determine
interphase
crystallography at
nanometer-scale
resolution*



Key techniques:

- Diffraction-Contrast STEM
- Scanning nano-beam diffraction
- Atomic-resolution STEM

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Orientations and phases in shear-bands can be distinguished through nanobeam diffraction

Austenite:

face-centered cubic (fcc)

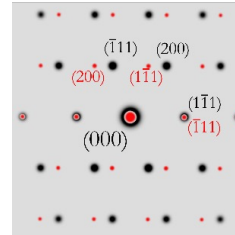
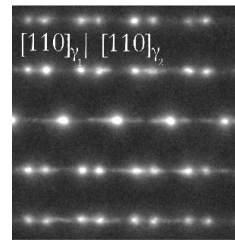
ϵ -martensite:

hexagonal close packed (hcp) structure

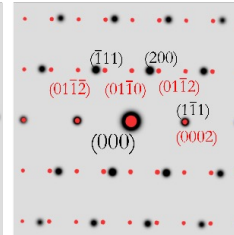
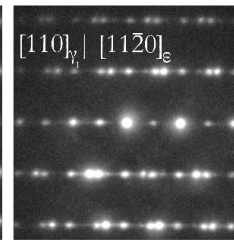
α' -martensite:

body-centered cubic (bcc)
(or bct depending on C-content)

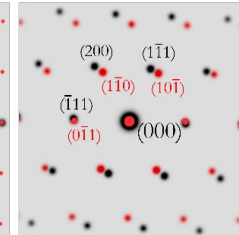
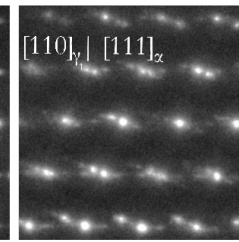
Austenite:
matrix & **twin**



Austenite &
 ϵ -martensite



Austenite &
 α' -martensite



Orientations align close-packed planes and directions:

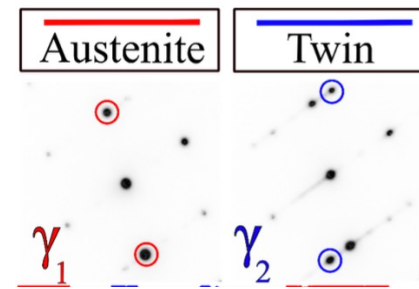
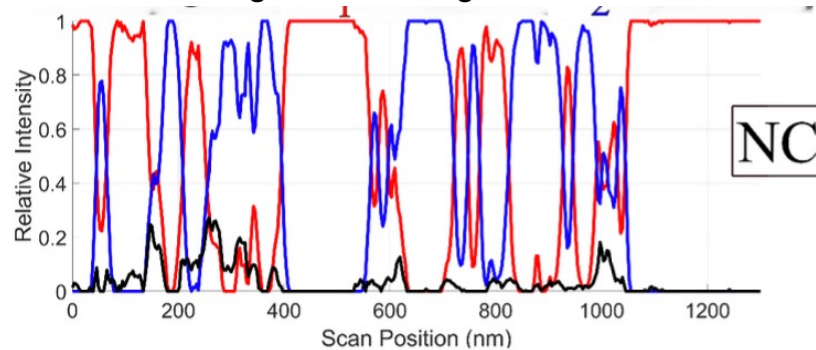
Austenite// ϵ -martensite: Burgers relation

Austenite// α' -martensite: Kurdumow-Sachs relation

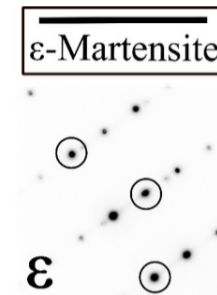
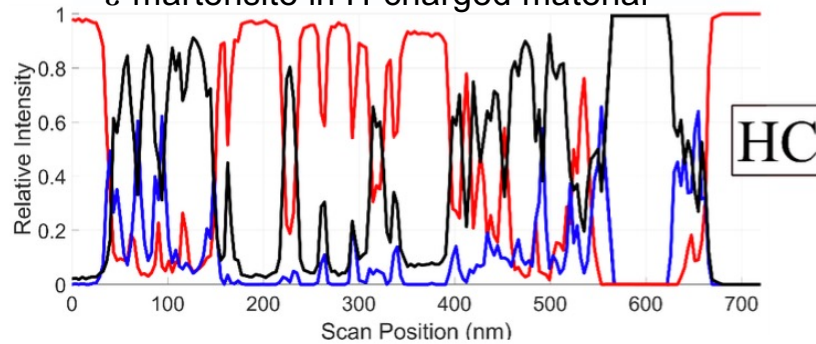
Hydrogen charging promotes ϵ -martensite formation in shear bands

Nanobeam diffraction line-scan analysis

Twining in non-charged material



ϵ -martensite in H-charged material

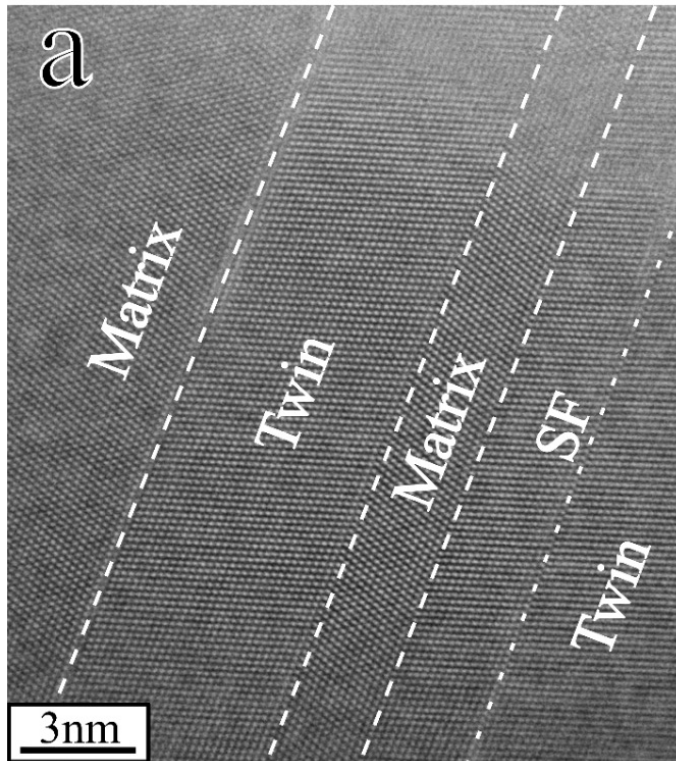


5% tensile strain

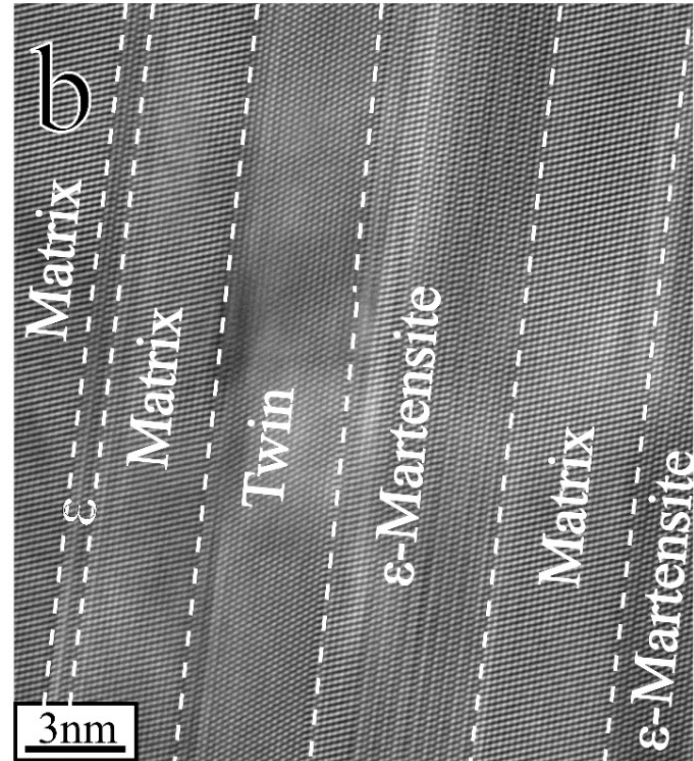
HRSTEM: detail of deformation bands

5% strain

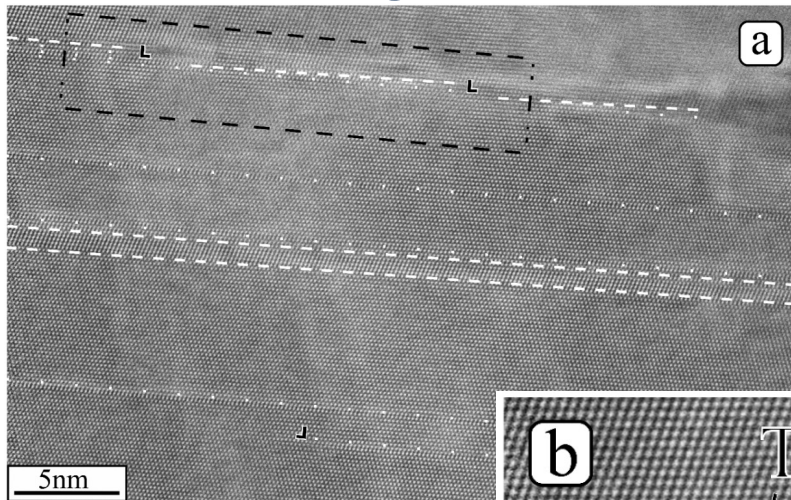
Non-Charged



Hydrogen-Charged

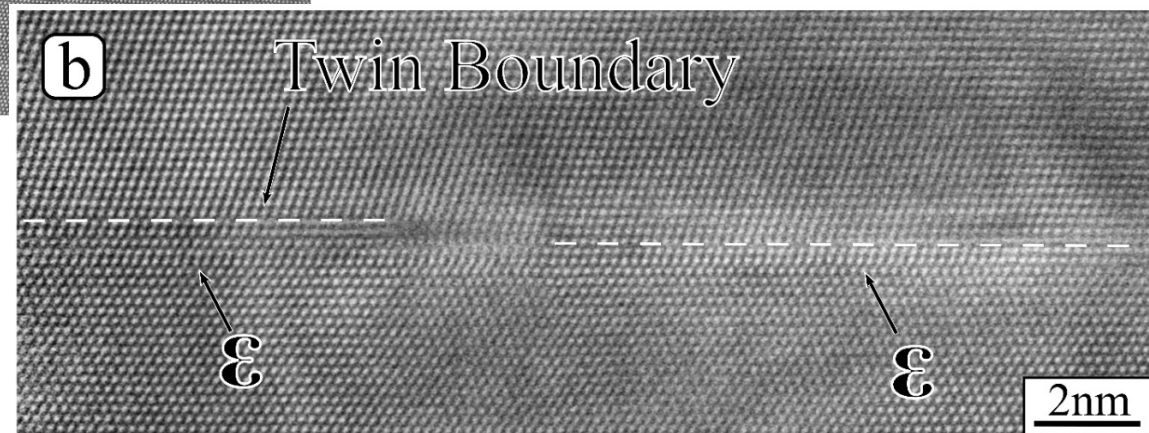


20% Strain: only very limited initiation of ϵ -martensite in non-charged material



Faulting in vicinity of twin boundaries

1-2 layers of ϵ -martensite



Promotion of ϵ -martensite by hydrogen: an open mechanistic question

Understanding relationship of H to stacking fault formation and partial dislocation motion is critical

-Reduction in SFE is often invoked as explanation H-influence on shear localization.

Existing experimental literature shows small reductions in SFE with H.

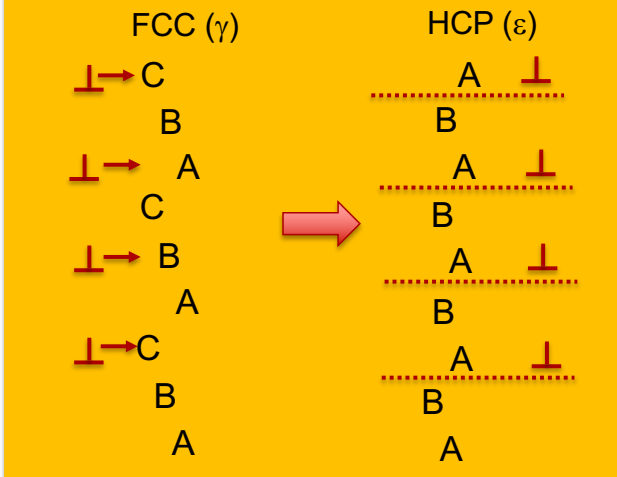
(e.g., Ferreira, Mat Sci Forum 1996, Pontini, Scripta Mat 1997)

-Solute drag effects:

Preferential pinning of trailing partials by hydrogen gives kinetic mechanism for fault extension. (e.g., Sills et al., 2016)

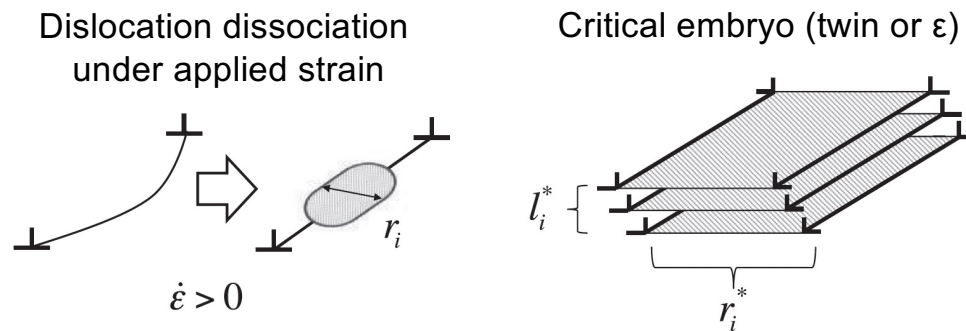
-Does not explain how faults would order into required ABAB... stacking sequence

FCC to HCP transformation by passage of series of $(1/6)\langle 112 \rangle$ dislocations



Twinning and martensite nucleation kinetics: sensitive to Stacking Fault Energy (SFE)

Recent micro-mechanics model describes balance between
nucleating deformation twins and martensite

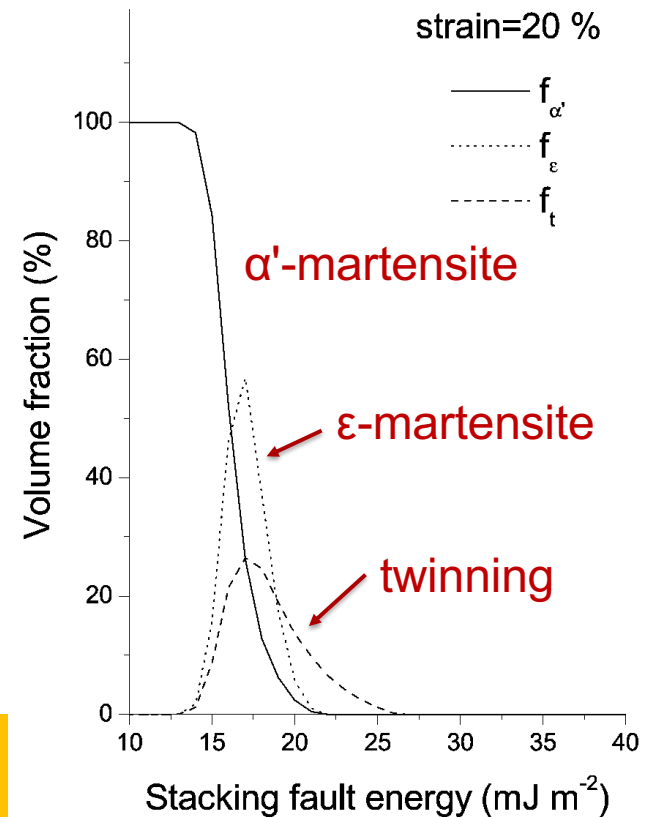


E.I. Galindo-Nava and P.E.J. Rivera-Díaz-del-Castillo,
Acta Mat. 128 (2017) 120-134

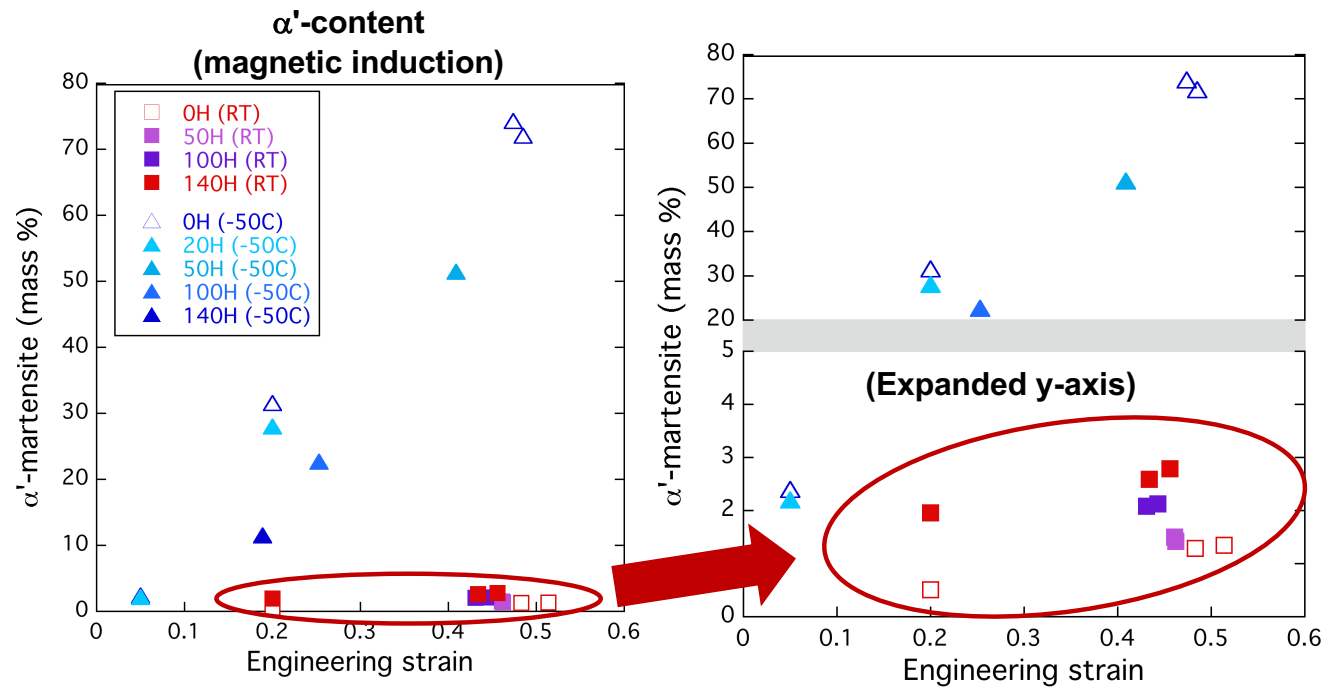
**Model predicts narrow region for which ϵ martensite
dominates over twinning.**

Dependent on SFE and loading conditions

Even small (few mJ/m^2) reductions in SFE from hydrogen may
be sufficient to transition from twinning to ϵ martensite.



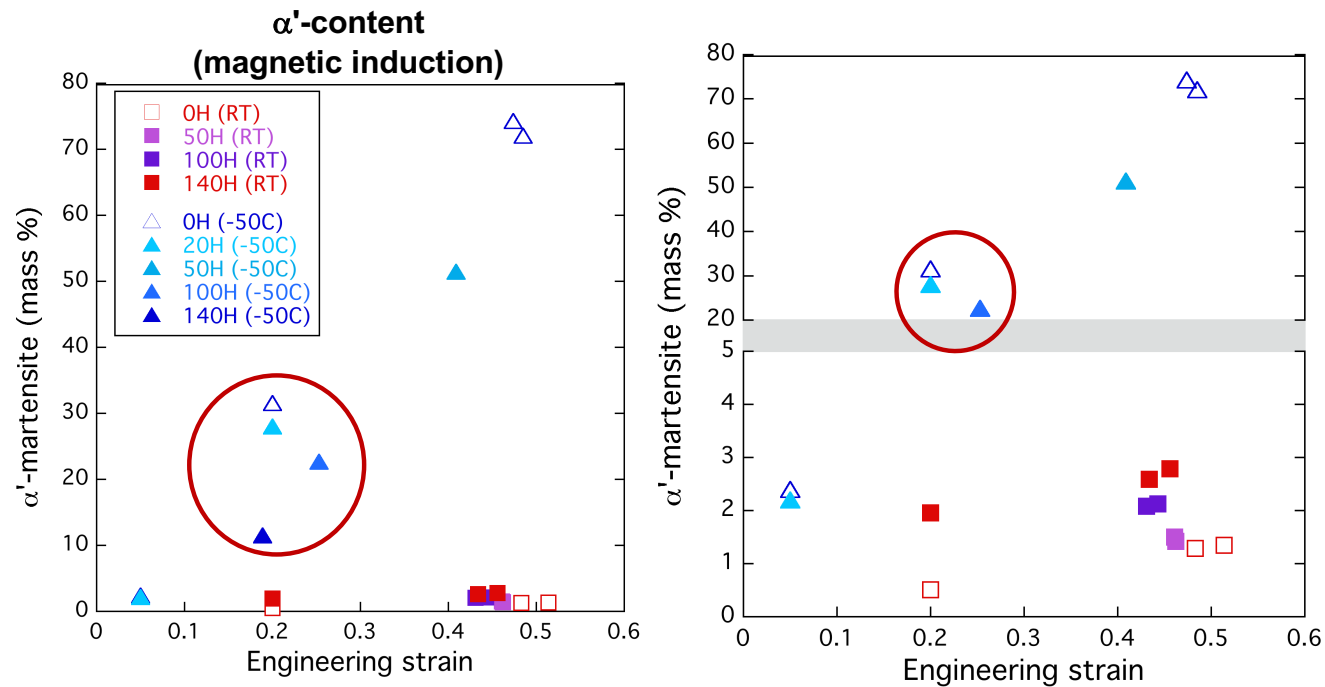
Influence of Hydrogen on Strain-induced α' -martensite



At **Room Temperature** and **low** volume of transformation,
hydrogen **promotes** α' -martensite

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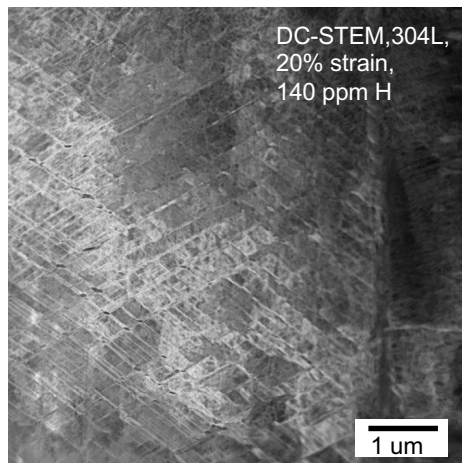
Influence of Hydrogen on Strain-induced α' -martensite



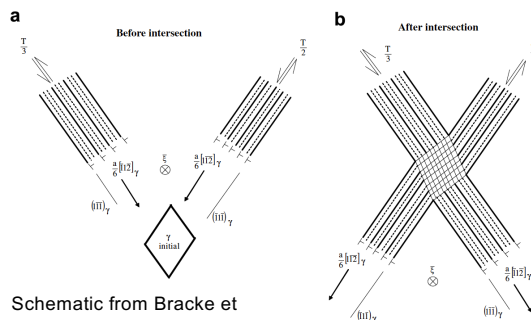
At **-50°C** and **high** volume of transformation,
hydrogen **suppresses** α' -martensite

C. San Marchi et al. International Journal of Hydrogen Energy 46 (2021)

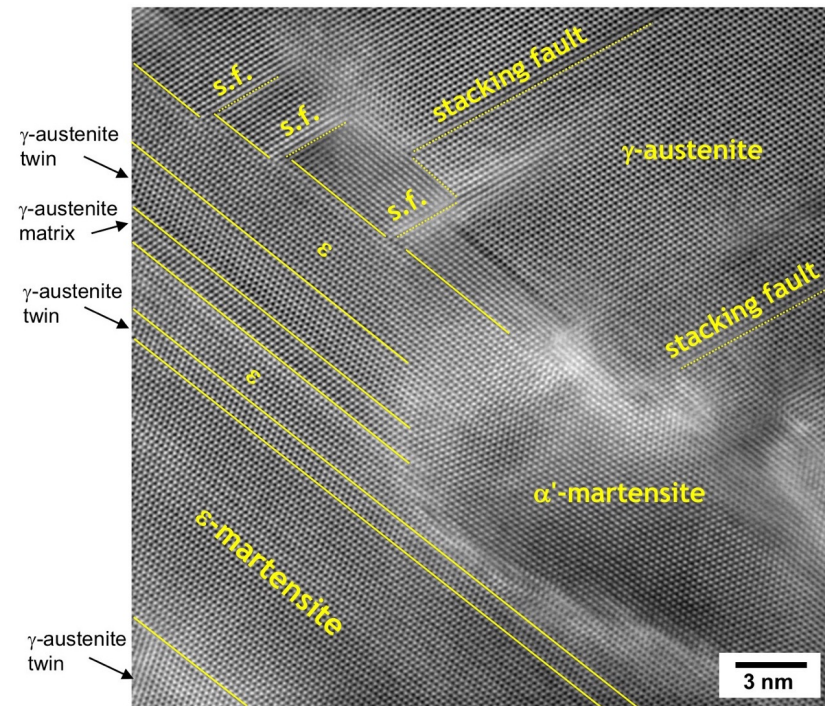
α' -martensite at shear band intersections: importance of ϵ -martensite



Olsen & Cohen model: α' -martensite nucleation at shear-band intersections with ϵ -martensite

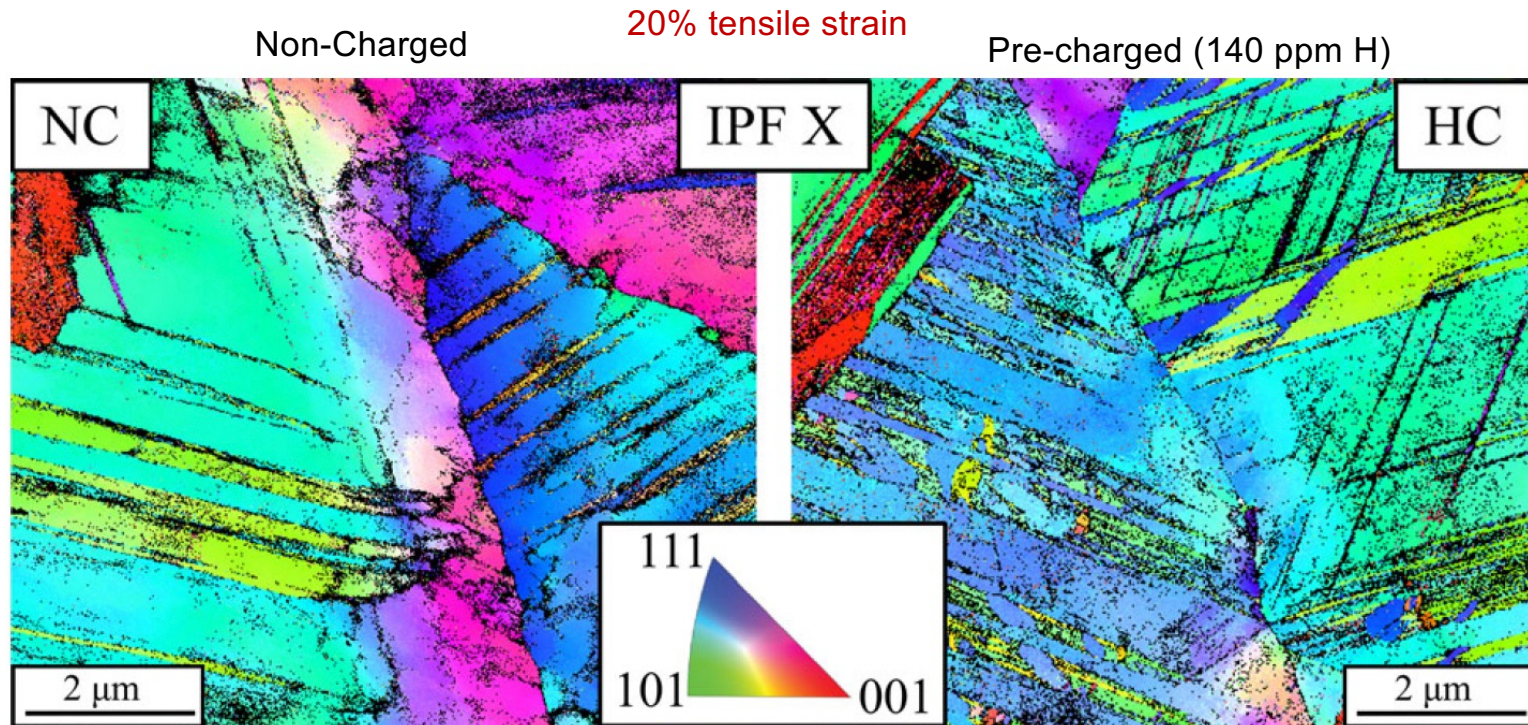


ϵ - and α' -martensite at shear bands in
tensile-strained 304L stainless steel
(20% strain, 140 ppm H)



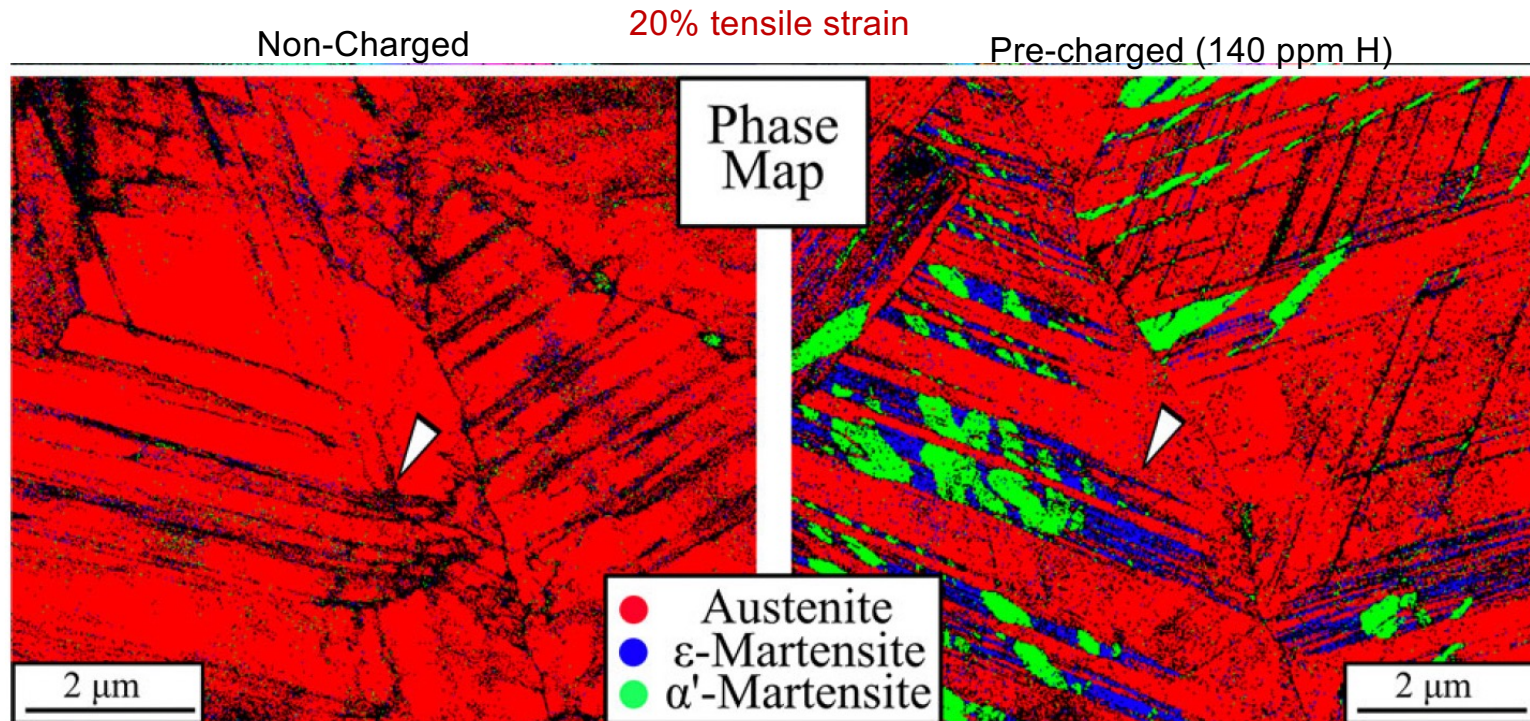
J.E.C. Sabisch et al. Metallurgical and Materials Transactions 2021

Colocation of ϵ and α' in hydrogen-charged 304L specimens



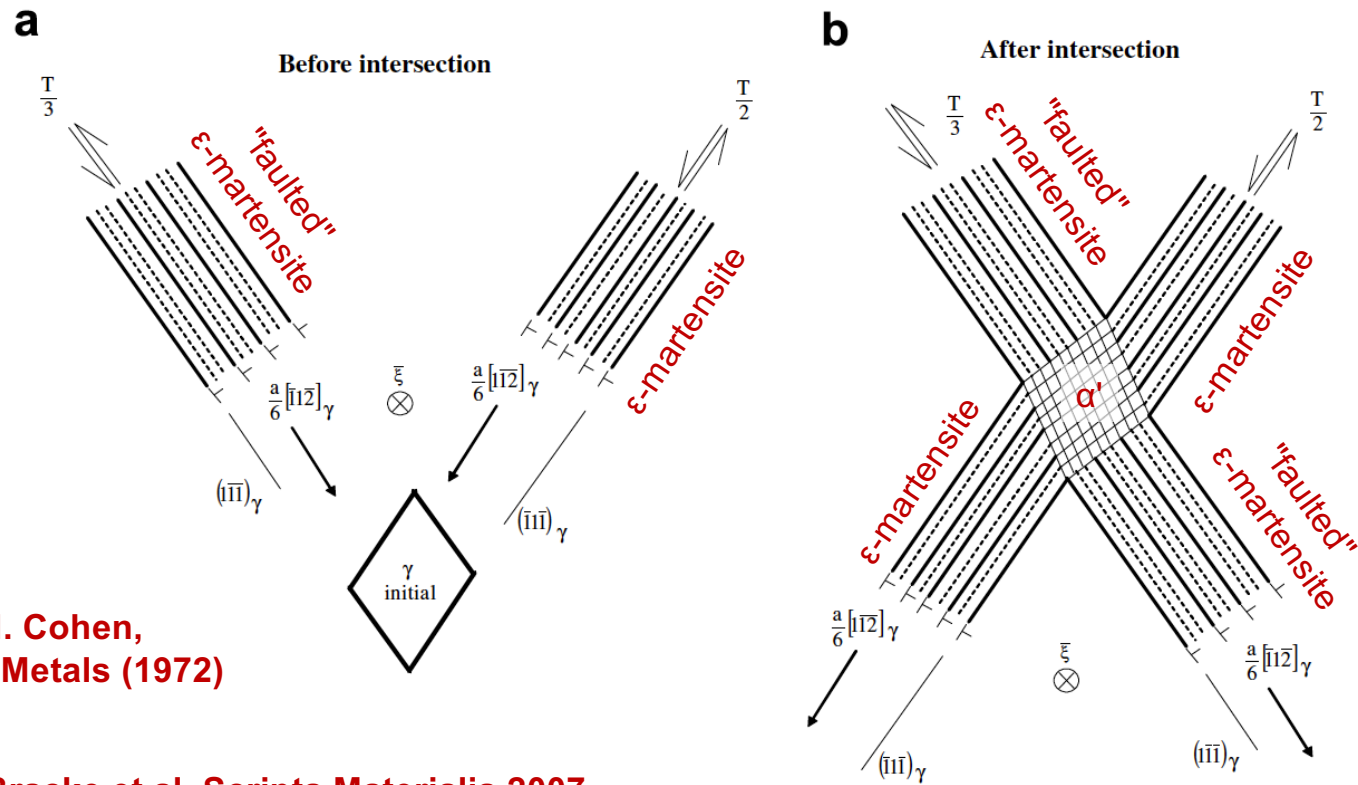
Transmission Kikuchi Diffraction (TKD) on electro-polished TEM specimens

Colocation of ϵ and α' in hydrogen-charged 304L specimens



Transmission Kikuchi Diffraction (TKD) on electro-polished TEM specimens

Olsen & Cohen model: α' -martensite nucleation at shear band intersections with ϵ -martensite

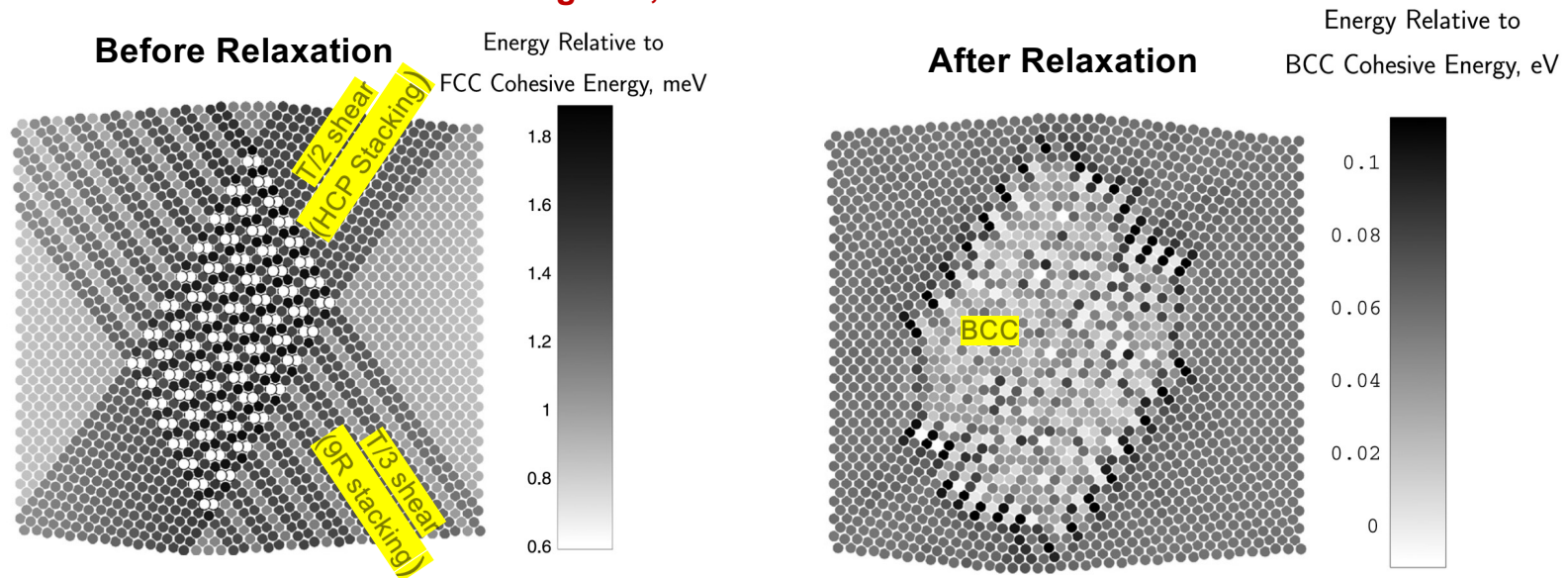


G.B. Olson and M. Cohen,
J. Less Common Metals (1972)

Schematic from Bracke et al. Scripta Materialia 2007

Atomistic calculations: α' nucleation at ε -martensite intersection

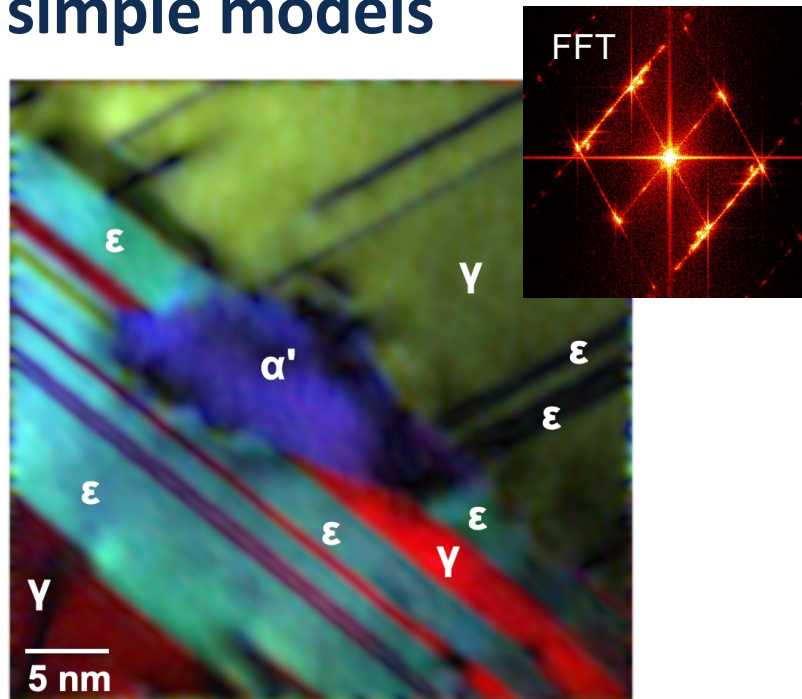
C.W. Sinclair and R.G. Hoagland, *Acta Materialia* 2008



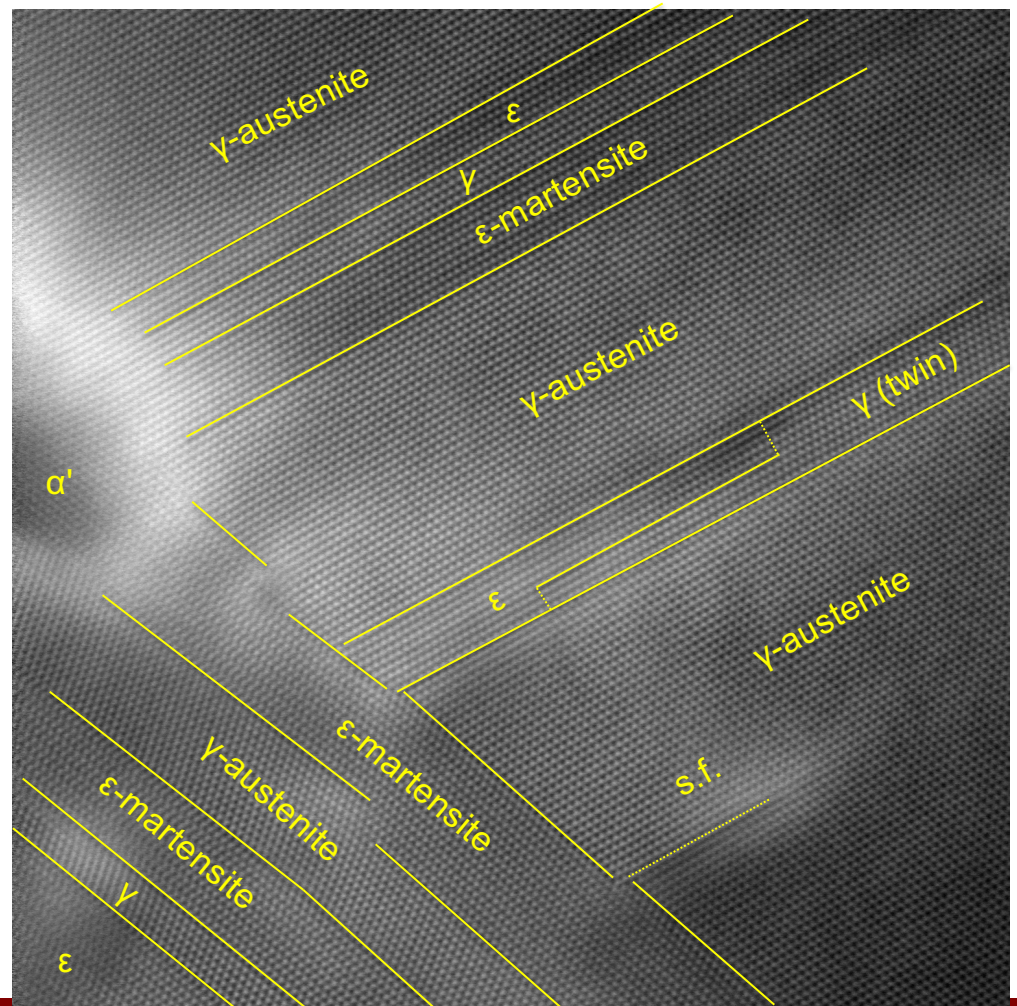
Increased potency for α' nucleation at intersections with ε -martensite

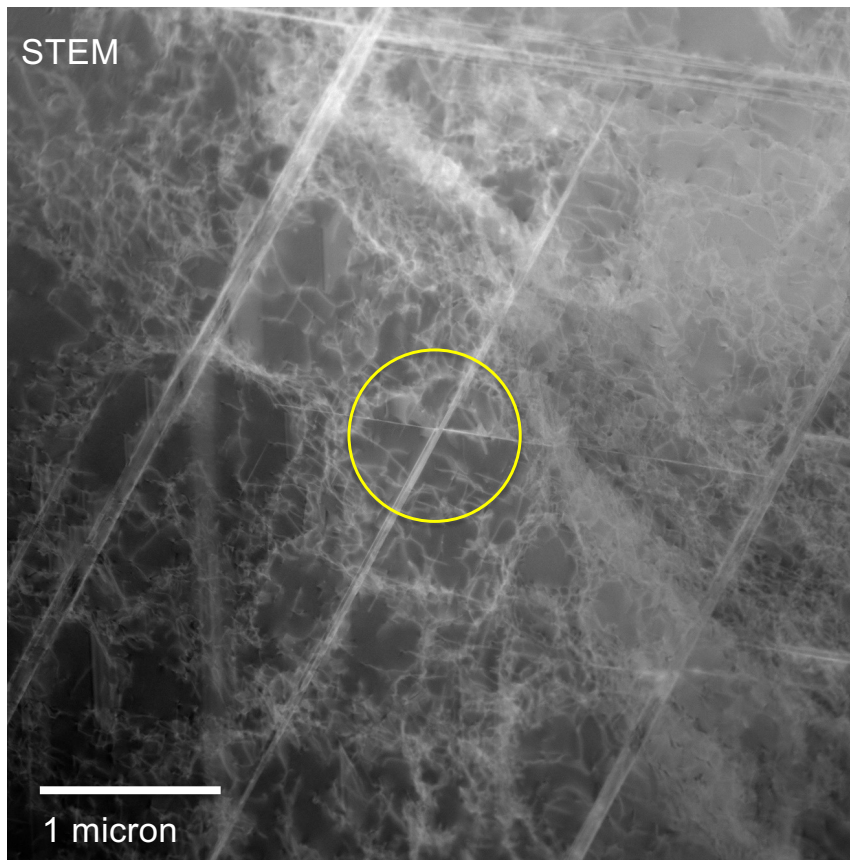
Increase of BCC α' -martensite at RT with internal hydrogen may be a secondary effect of hydrogen increasing ε -martensite formation

Shear Band Intersections: More complicated than simple models



Geometric Phase Analysis (GPA)
from HAADF-STEM





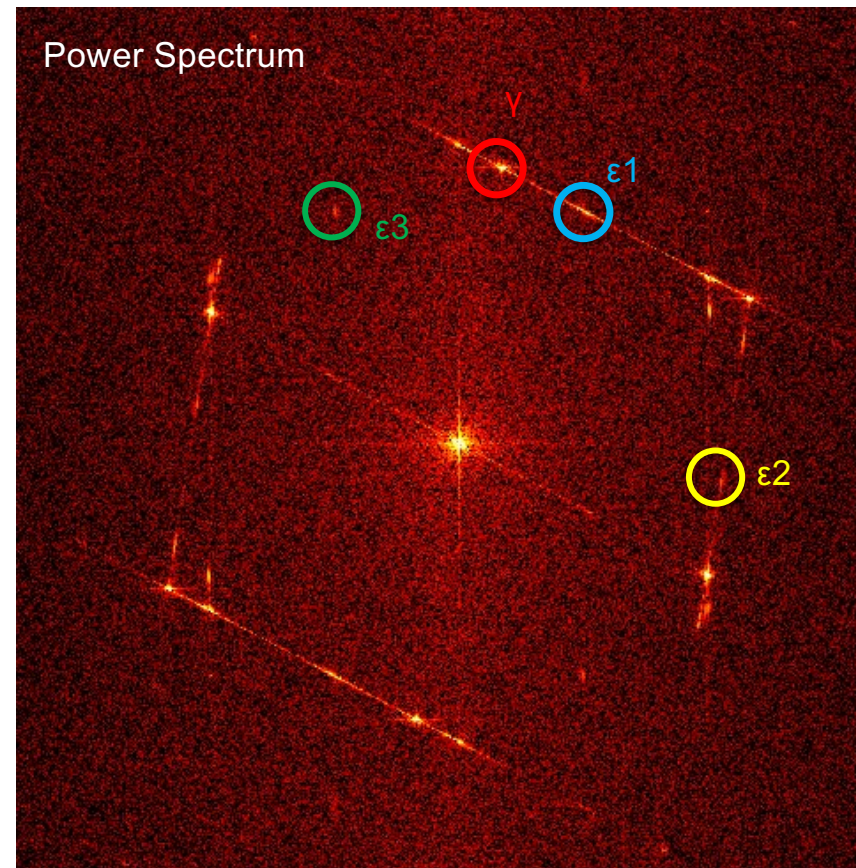
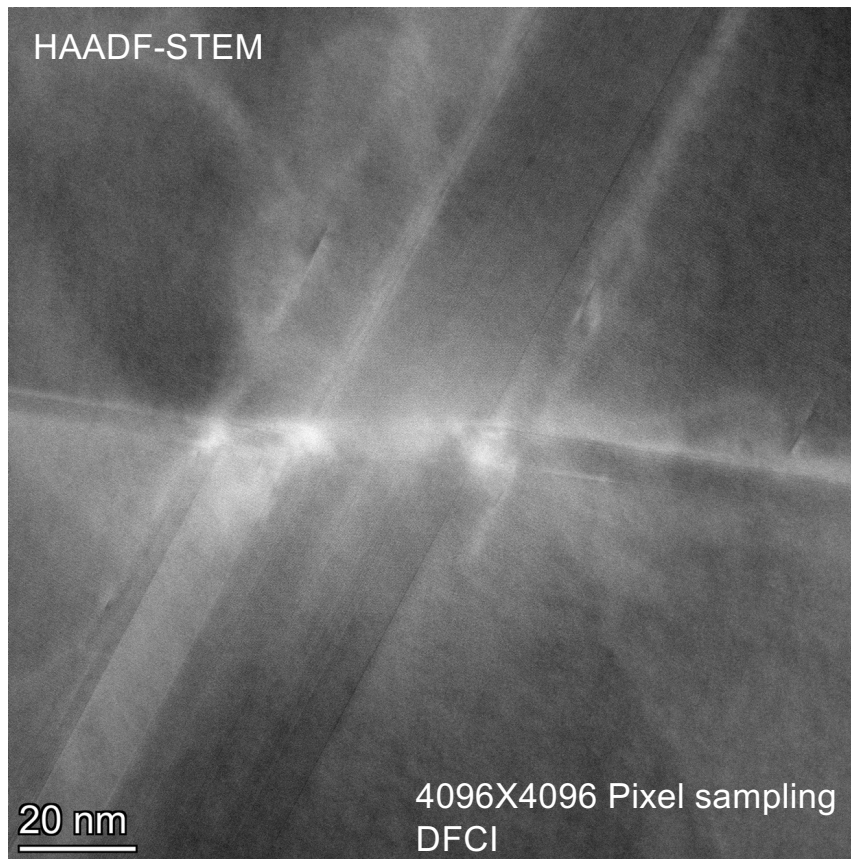
ϵ - ϵ intersections

An alternative to
nucleating α' martensite:

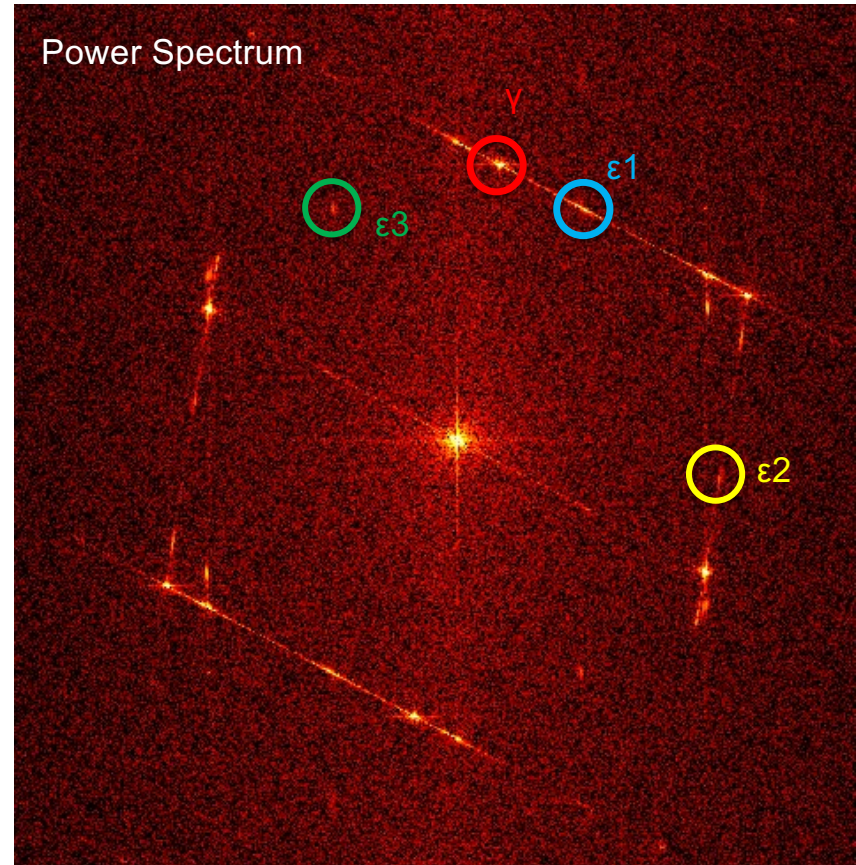
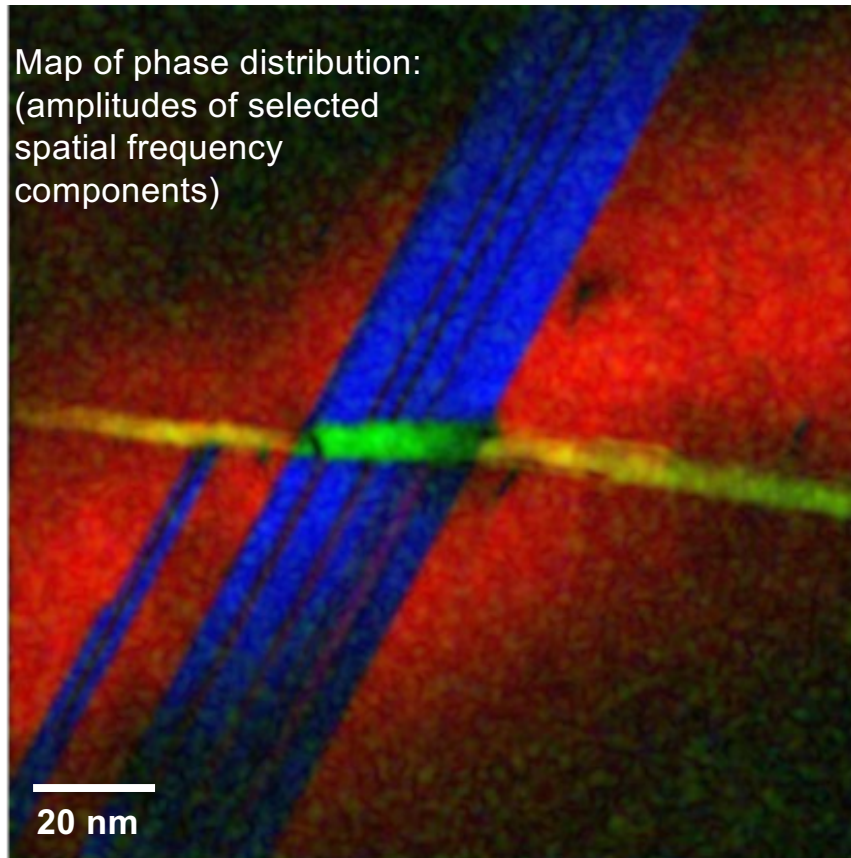
Twinning of the ϵ phase

304L, 5% strain, -50 C, 140 wppm H

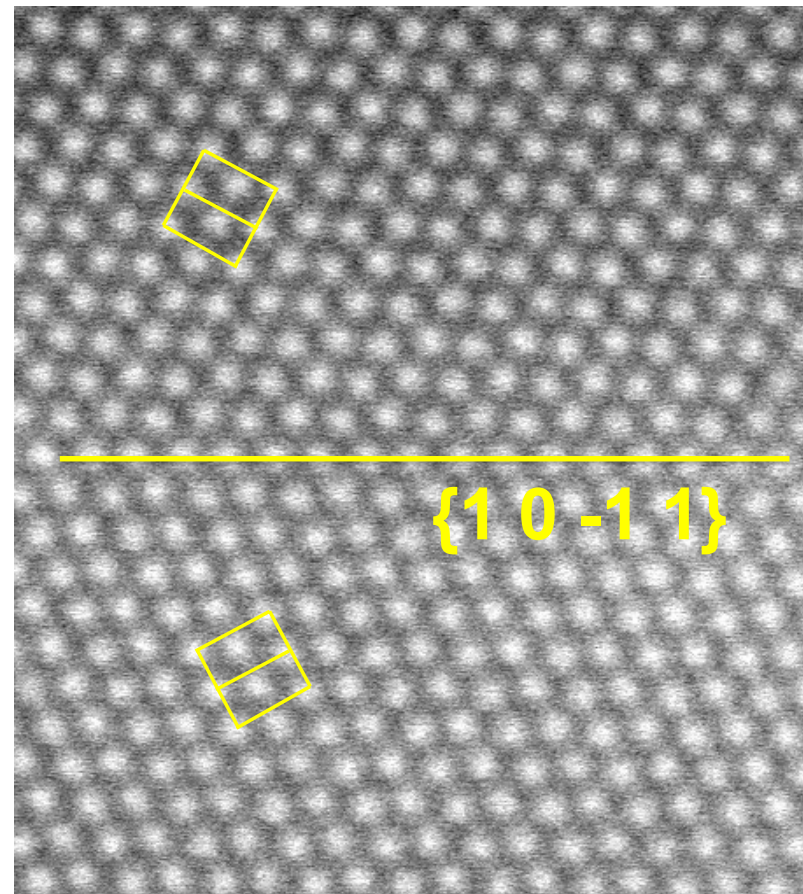
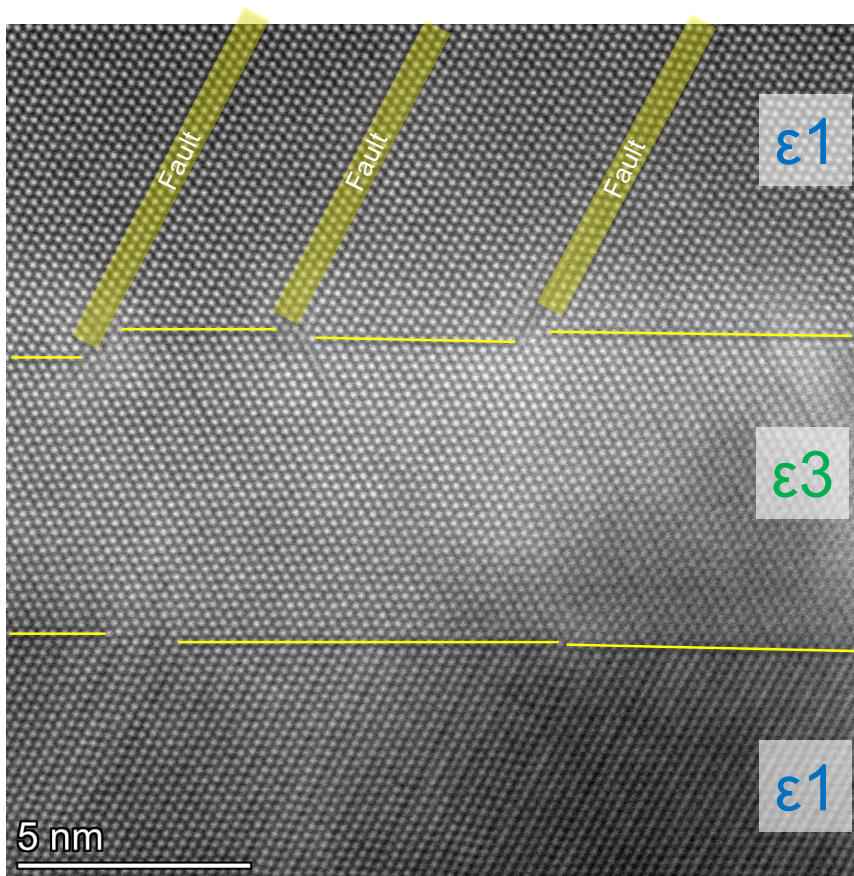
ε - ε intersection: Analysis of variants



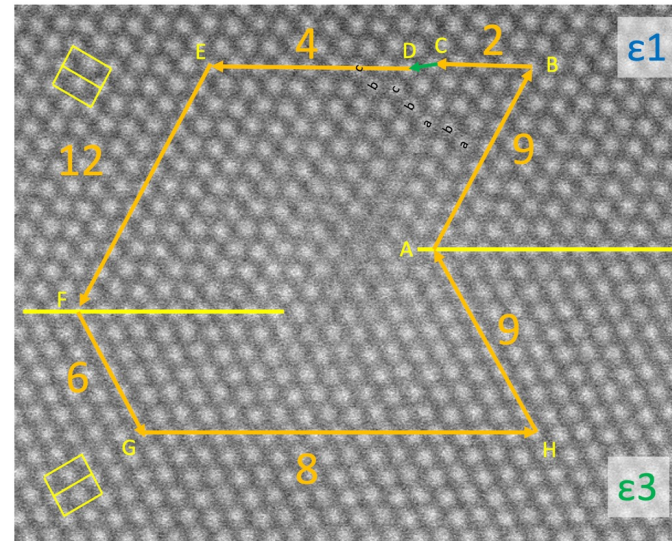
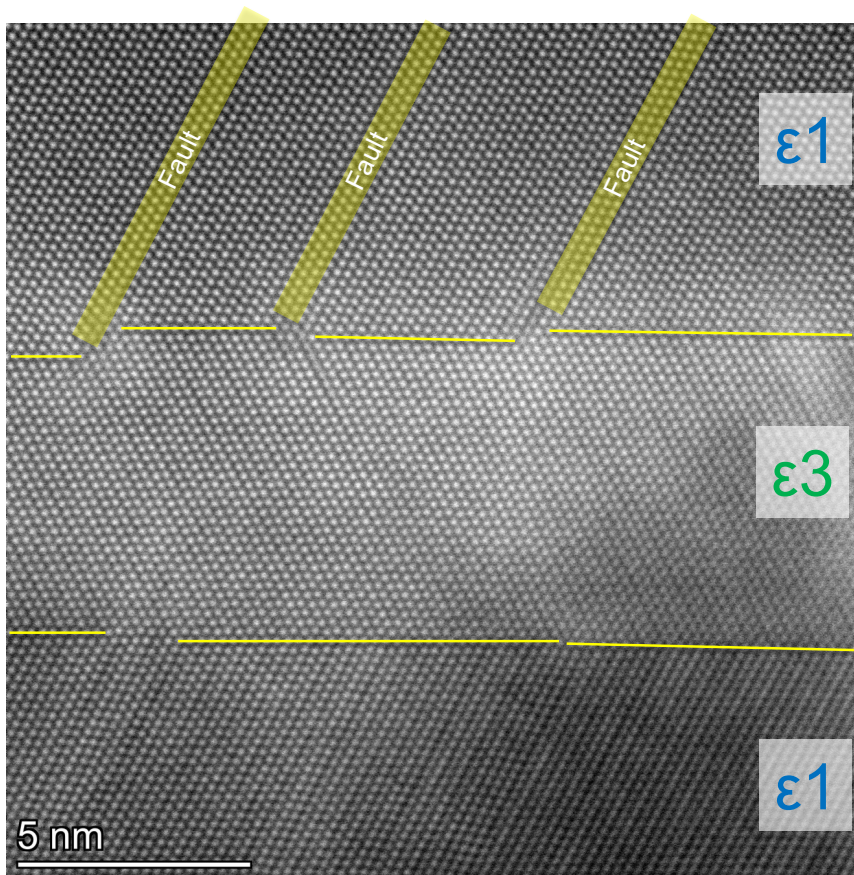
ϵ - ϵ intersection: Analysis of variants



ϵ - ϵ intersection: $\{1\ 0\ -1\ 1\}$ twinning



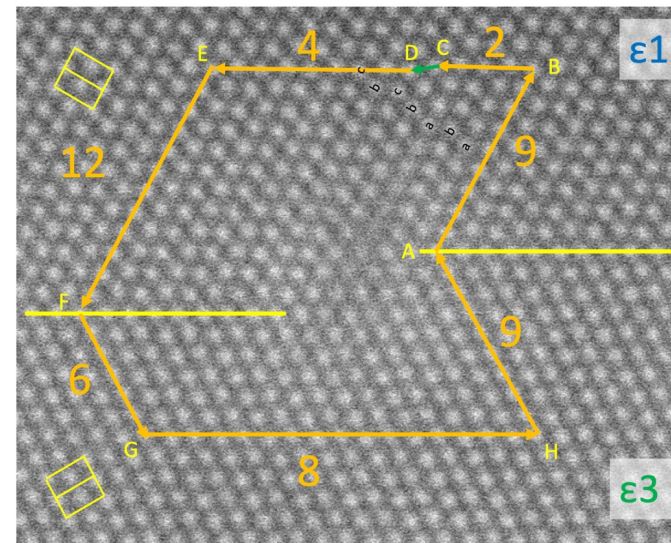
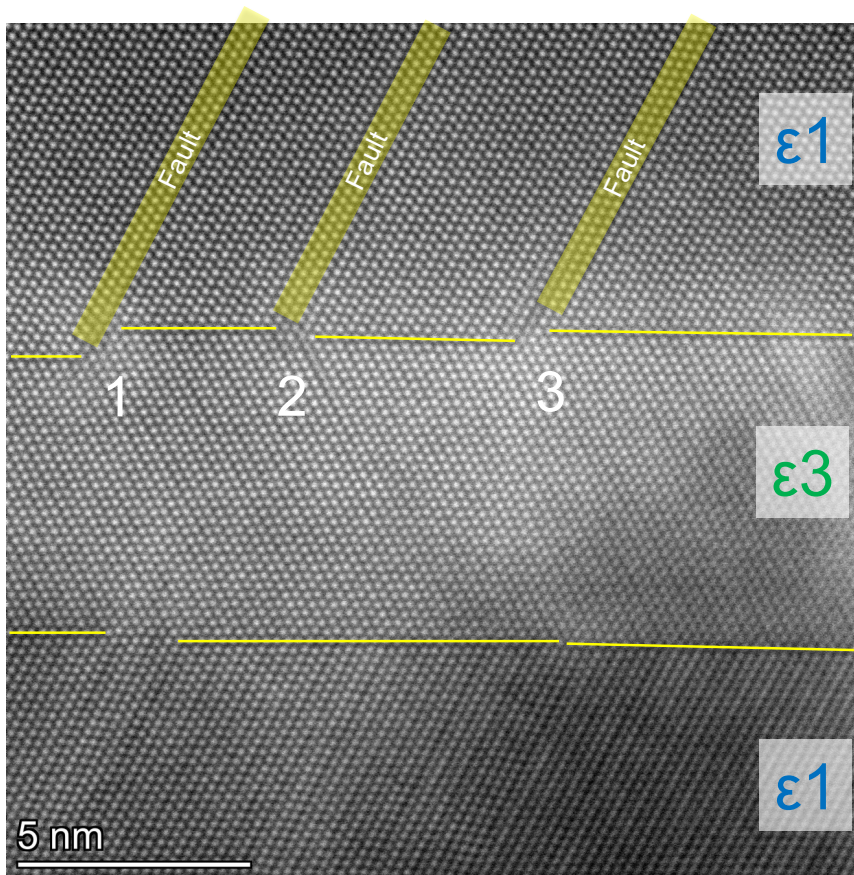
ϵ - ϵ intersection: $\{1\ 0\ -1\ 1\}$ twinning



Circuit Mapping: $\mathbf{b} = -(\mathbf{C}_\lambda + P\mathbf{C}_\mu)$

A. Serra, R.C. Pond, D.J. Bacon, Acta Mat 1991
 $\{10\text{-}11\}$ Twins in HCP

ϵ - ϵ intersection: $\{1\ 0\ -1\ 1\}$ twinning



	Step Heights ($d, \{1011\}$)		Burgers Vector
	$h_{\epsilon1}$	$h_{\epsilon2}$	(x,y,z) components, units of a
1	$+3\frac{1}{6}$	+3	(0, -0.2705, +1275)
2	$-\frac{5}{6}$	-1	(0, +0.1803, +1275)
3	$+\frac{5}{6}$	+1	(0, -0.1803, -1275)

Conclusions

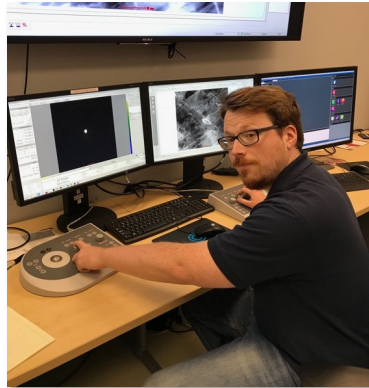
- Complex, multiscale evolution of microstructure under tensile strain in forged austenitic stainless steel.
 - Microstructure affected by presence of internal hydrogen
 - Strain localization into planar deformation bands
 - Twinning in non-charged 304L*
 - Both twinning and martensite formation in H-charged 304L*
 - ϵ –martensite in shear-bands
 - α' -martensite at intersections of shear-bands
 - ϵ –martensite provides a favorable pathway to α' .
 - Likely that the initial increase in α' with H is a secondary effect of hydrogen promoting ϵ -martensite formation, aiding α' nucleation*
 - Ongoing work:
 - Promotion of ϵ -martensite formation by H?*
 - Low T reduction of α' by presence of H?*
 - Detailed atomistic processes at shear-band intersections*

Special Acknowledgments

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Sandia National Labs



Recent Papers

J.E.C. Sabisch, J.D. Sugar, J.A. Ronevich, C. San Marchi, D.L. Medlin, "Interrogating the Effects of Hydrogen on the Behavior of Planar Deformation Bands in Austenitic Stainless Steel", Metallurgical and Materials Transactions A. 52, 1516-1525 (2021): <https://doi.org/10.1007/s11661-021-06170-3>.

C. San Marchi, J.A. Ronevich, J.E.C. Sabisch, J.D. Sugar, D.L. Medlin, B.P. Somerday, "Effect of microstructural and environmental variables on ductility of austenitic stainless steels". International Journal of Hydrogen Energy, 46 (2021) 12338-12347. <https://doi.org/10.1016/j.ijhydene.2020.09.069>