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Subcritical Crack Growth in High-pressure Hydrogen and Hydrogen With Oxygen Impurity

Robert W. Wheeler, Chris San Marchi, and Joseph Ronevich

Hydrogen and Materials Science Department, Sandia
National Laboratories Livermore, CA, USA

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Overview

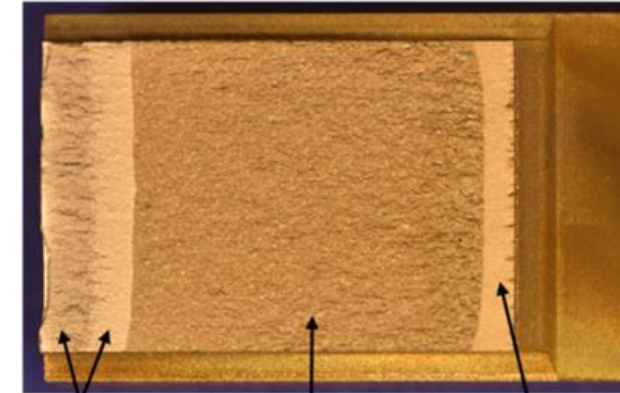
- Background and Motivation
- Experimental Methods
- Subcritical Crack Growth in High-pressure Hydrogen and Hydrogen With Oxygen Impurity
 - SA372 Grade J steel
 - SA372 Grade L steel
 - X100 pipeline steel
 - 13-8 stainless steel
- Summary and Conclusions

Wedge-opened Loaded (WOL) Sample



Reaction pin (load tup) with externally monitored strain gauge

Grade L: WOL Fracture Surface

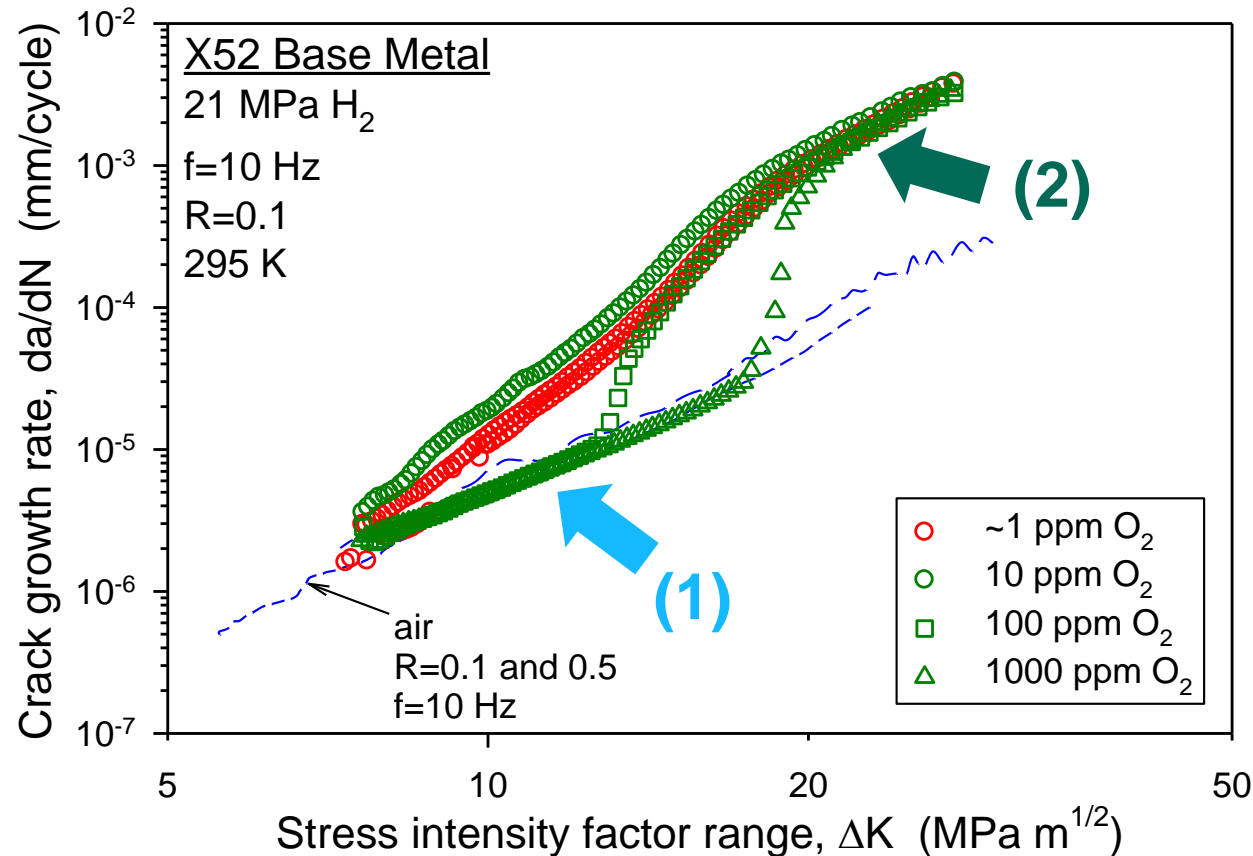


Post-test fatigue and LN_2 fracture

Fatigue precrack (in air)

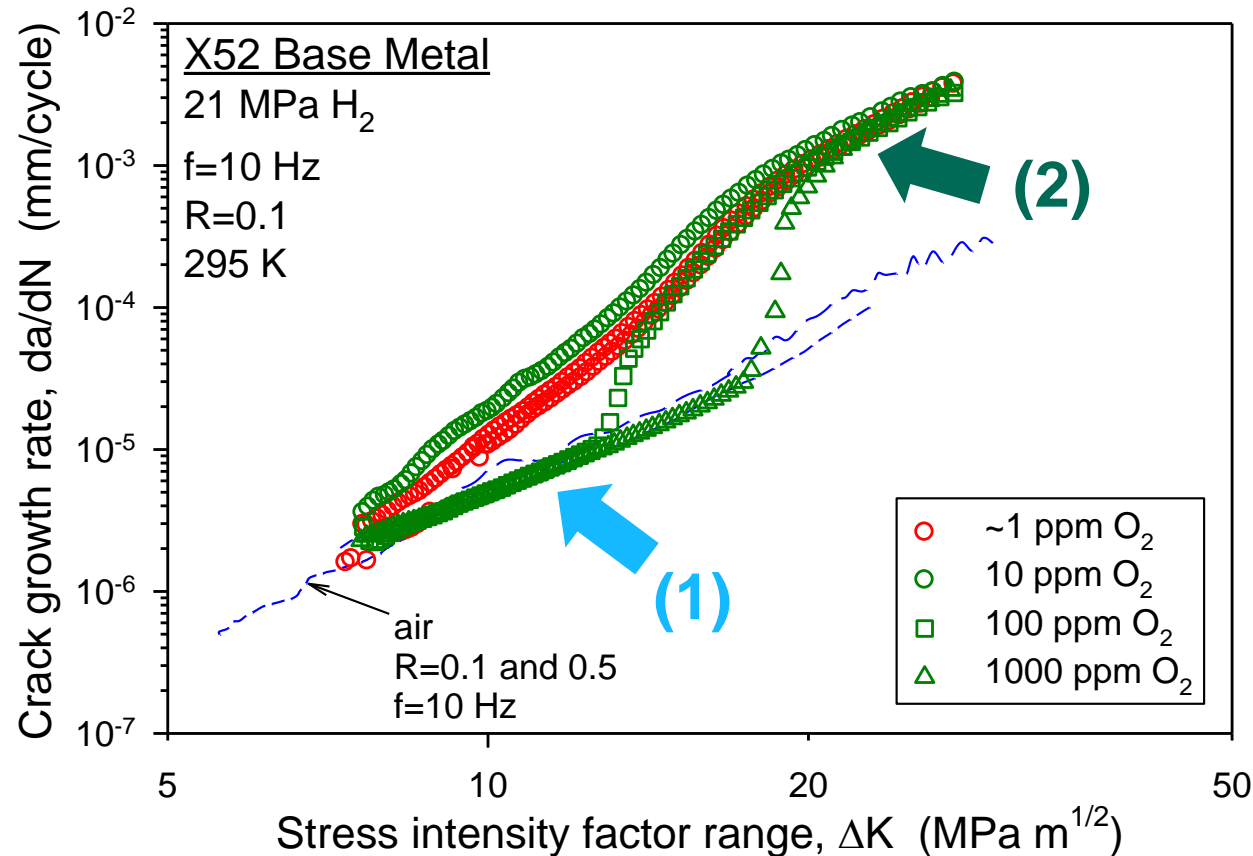
Crack growth in H_2 + 100PPM O_2

Oxygen is known to affect measurements of fatigue and fracture



- Numerous examples of trace gases mitigating fatigue crack growth rate (FCGR) in laboratory conditions
- Example:
 - (1) Oxygen reduces FCGR to air
 - (2) Oxygen has no effect on FCGR in H₂

Oxygen is known to affect measurements of fatigue and fracture



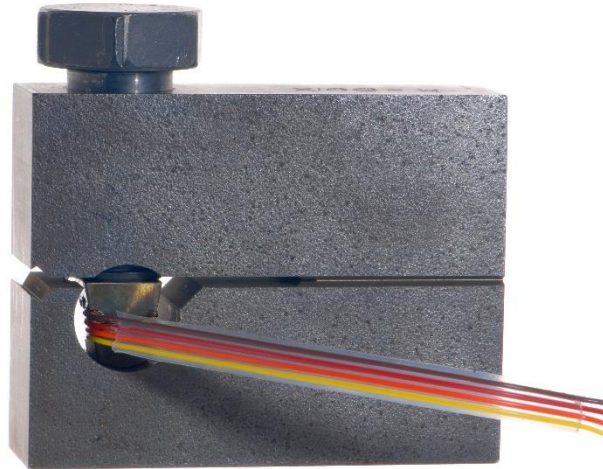
- Fatigue crack growth tests are typically performed at 1 Hz (\pm decade)
 - $da/dN = 10^{-5}$ mm/cycle
 - Time for $\Delta a = 1$ mm: ~ 1 day
 - 1 day = 0.02% of 10 year life
- Are the time scales of a typical laboratory fatigue test sufficient to demonstrate kinetics over decades?
 - More accurately simulate the mechanical/environmental conditions that components see when in use
 - Does trace oxygen have long term mitigation effects on hydrogen embrittlement?

Sustained load testing can be executed over periods of days to weeks to months to years



- Fixed displacement tests
- Placed in pressure vessels & pressurized up to 140 MPa gaseous environment
 - Experiments in this study were performed at 103MPa
- Test durations can range from days to years
- Instrumented reaction pins allows us to determine incubation time
- Directly compare subcritical crack growth in hydrogen and mixed gas environments

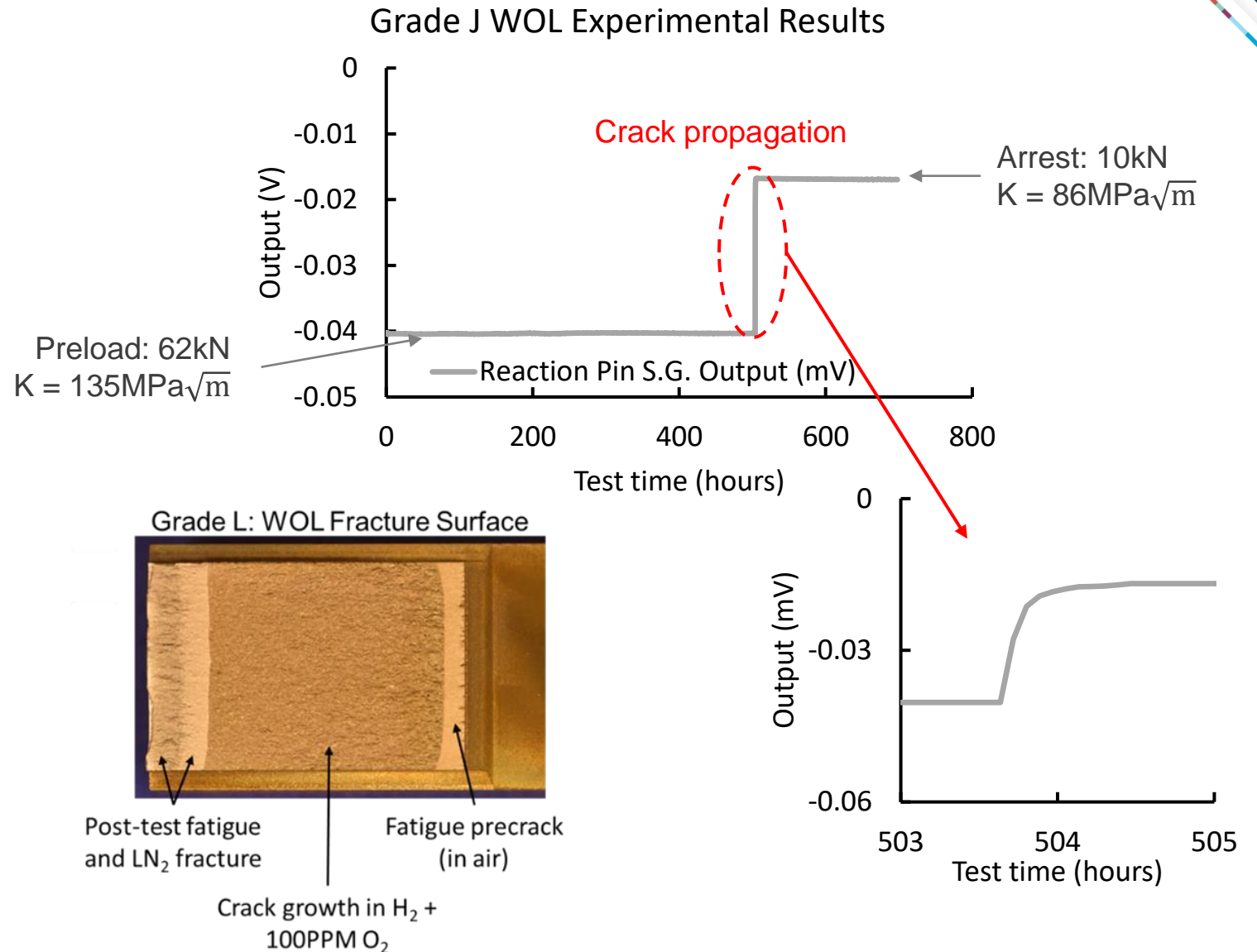
Wedge-opened loaded (WOL)



ASTM E1681 – Threshold Stress Intensity Factor for Environment-Assisted Cracking

Crack initiation and growth rates can be measured during constant displacement fracture experiments

- Instrumented reaction pins allow for determination of incubation time and crack growth rates
 - Continuous data collection throughout the duration of the experiments
- Time between the initial crack propagation and arrest can range between seconds to hours
 - With a constant displacement, the crack growth rates can be determined from the load on the reaction pin
- Post-test fatigue and heat tinting are used to mark fracture surfaces



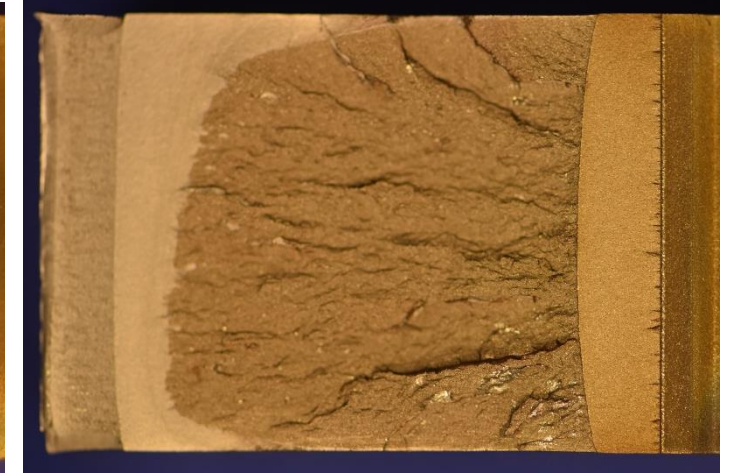
Material selection and fracture surfaces

- SA372 Grade J steel
 - YS = 700 MPa
- SA372 Grade L steel
 - YS = 730 MPa
- X100 pipeline steel
 - YS = 760 MPa
- Precipitation Hardened 13-8 stainless steel
 - YS = 1480 MPa

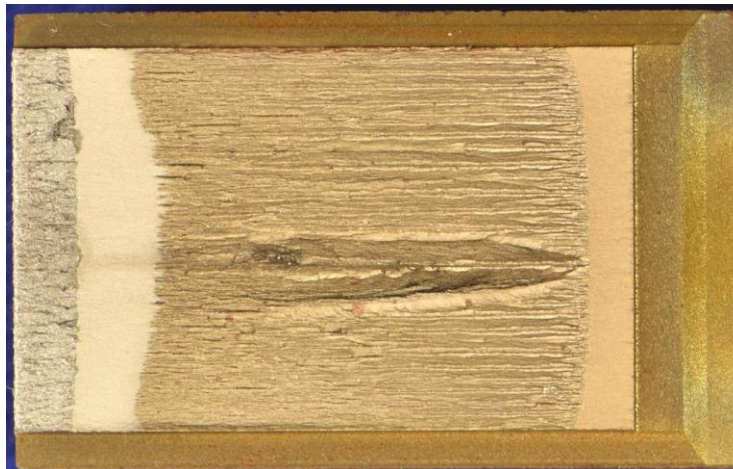
Grade L



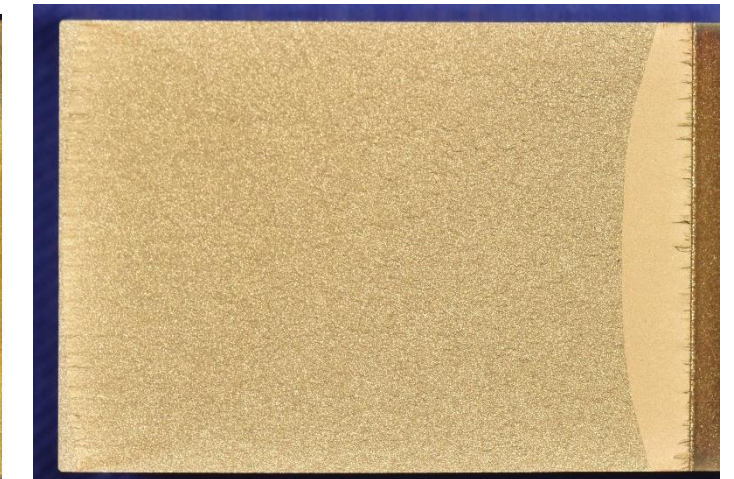
Grade J



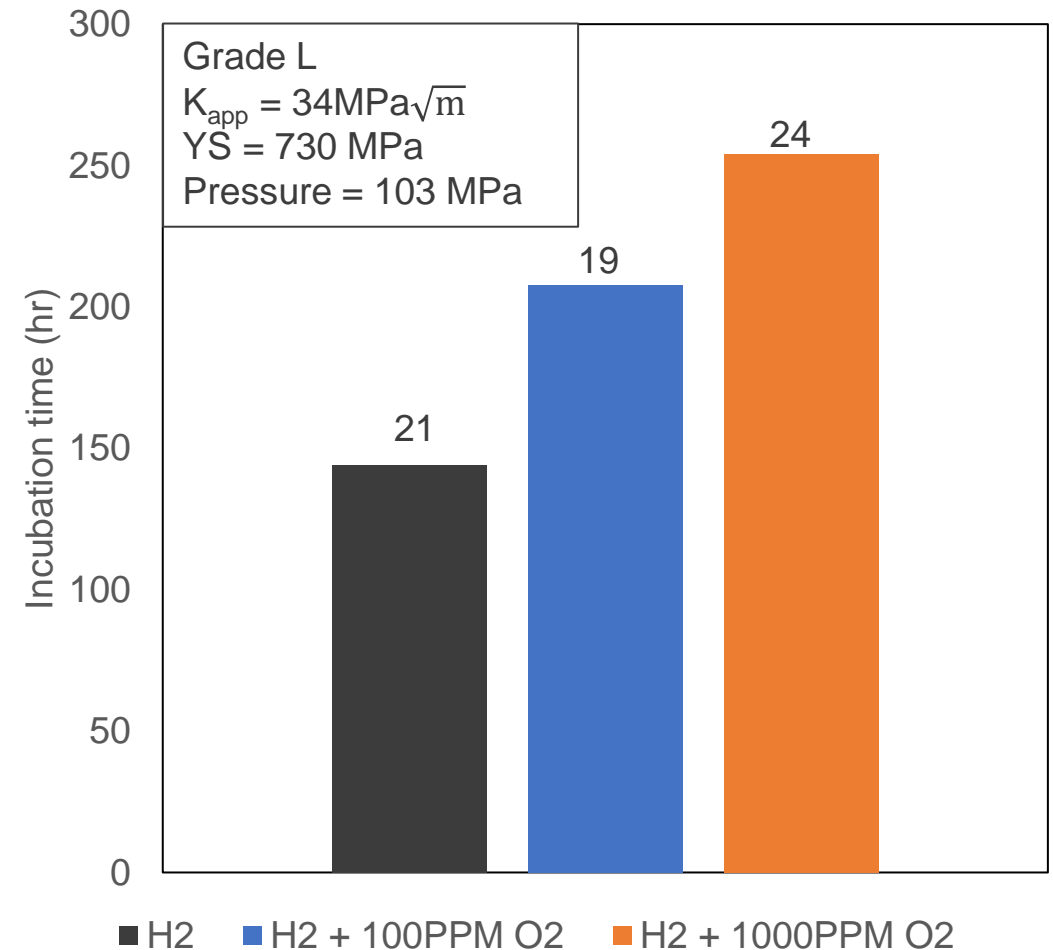
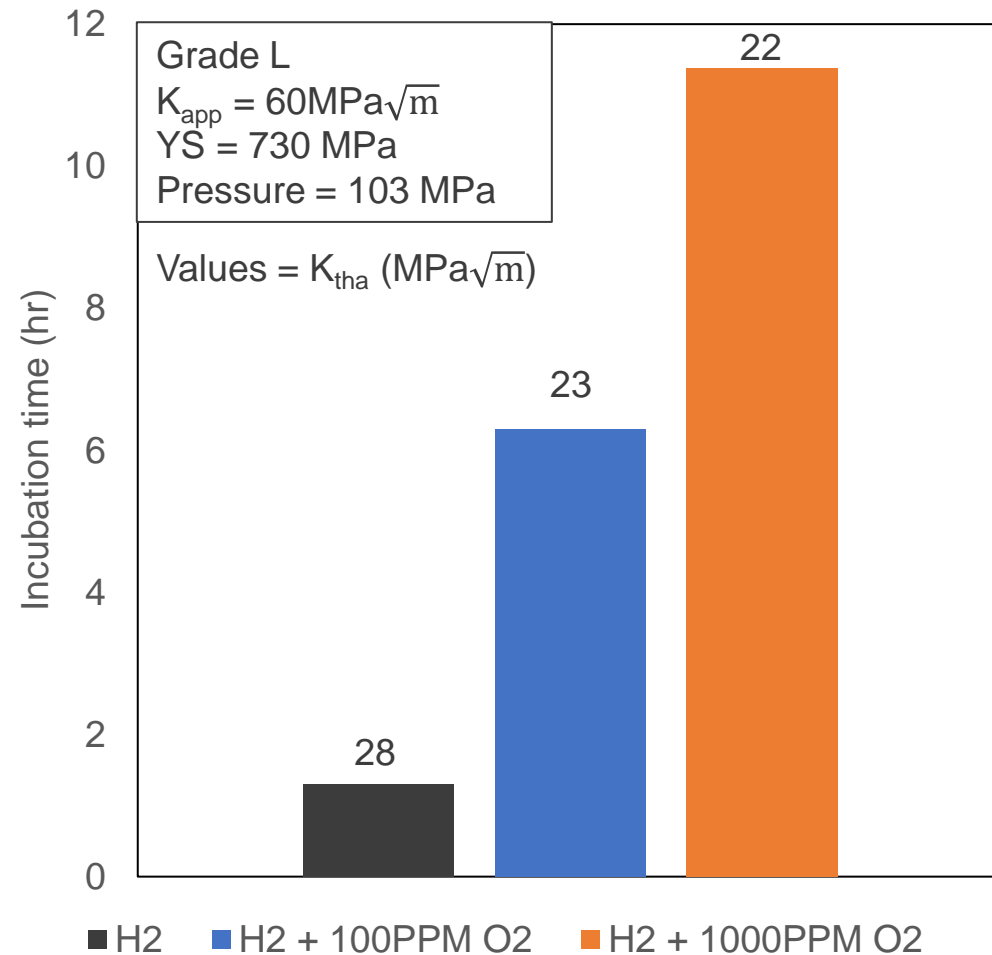
X100



13-8

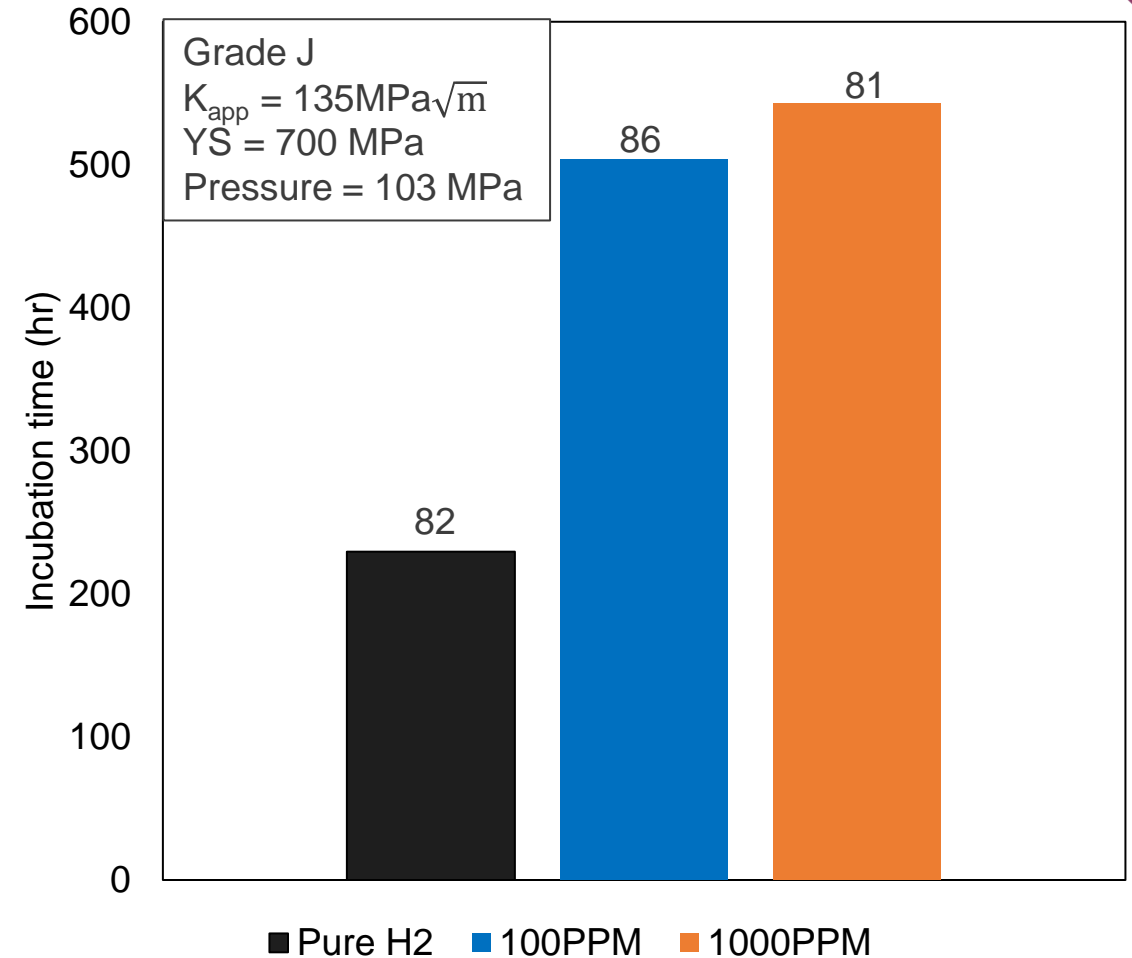
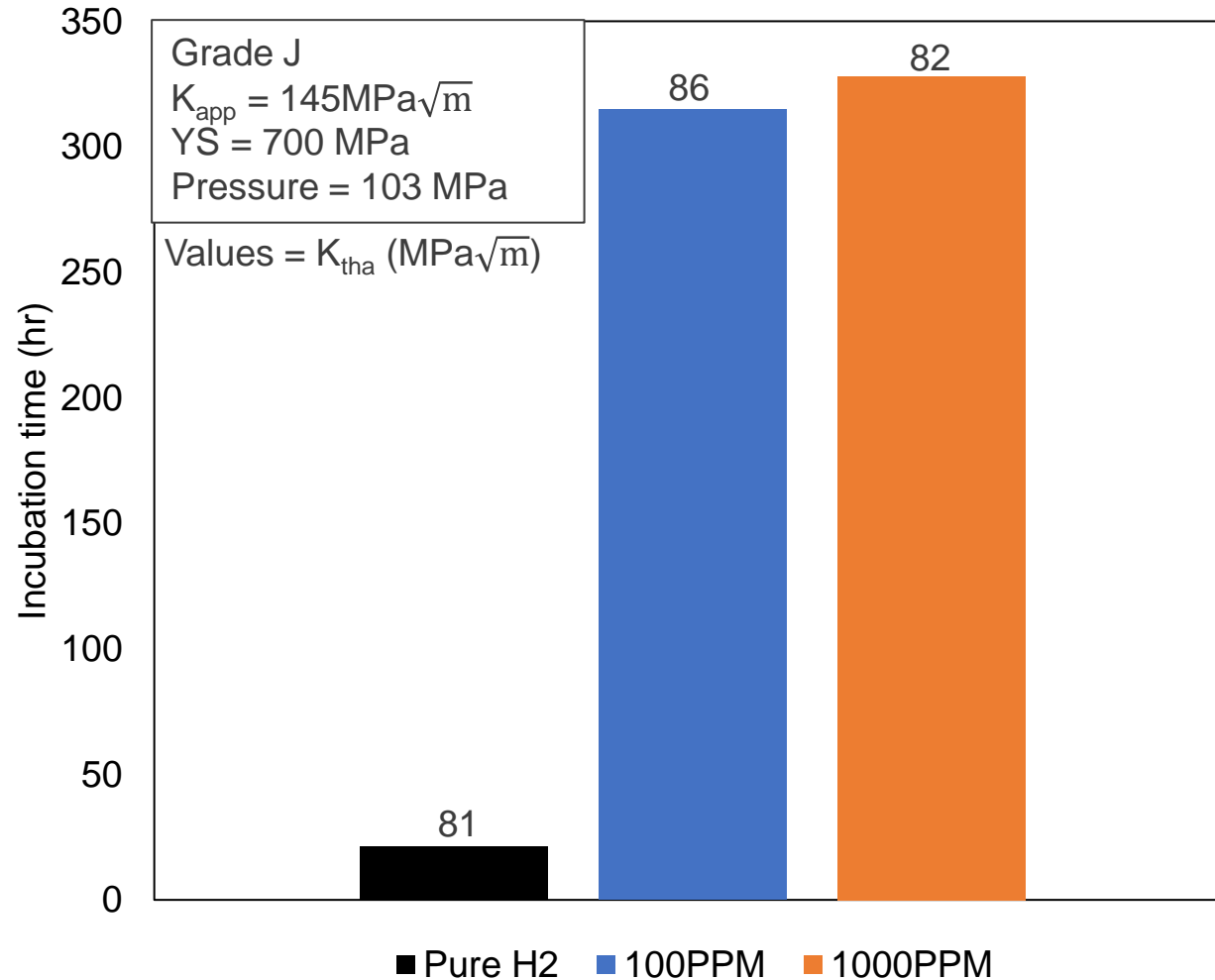


100PPM & 1000PPM O₂ delay incubation time



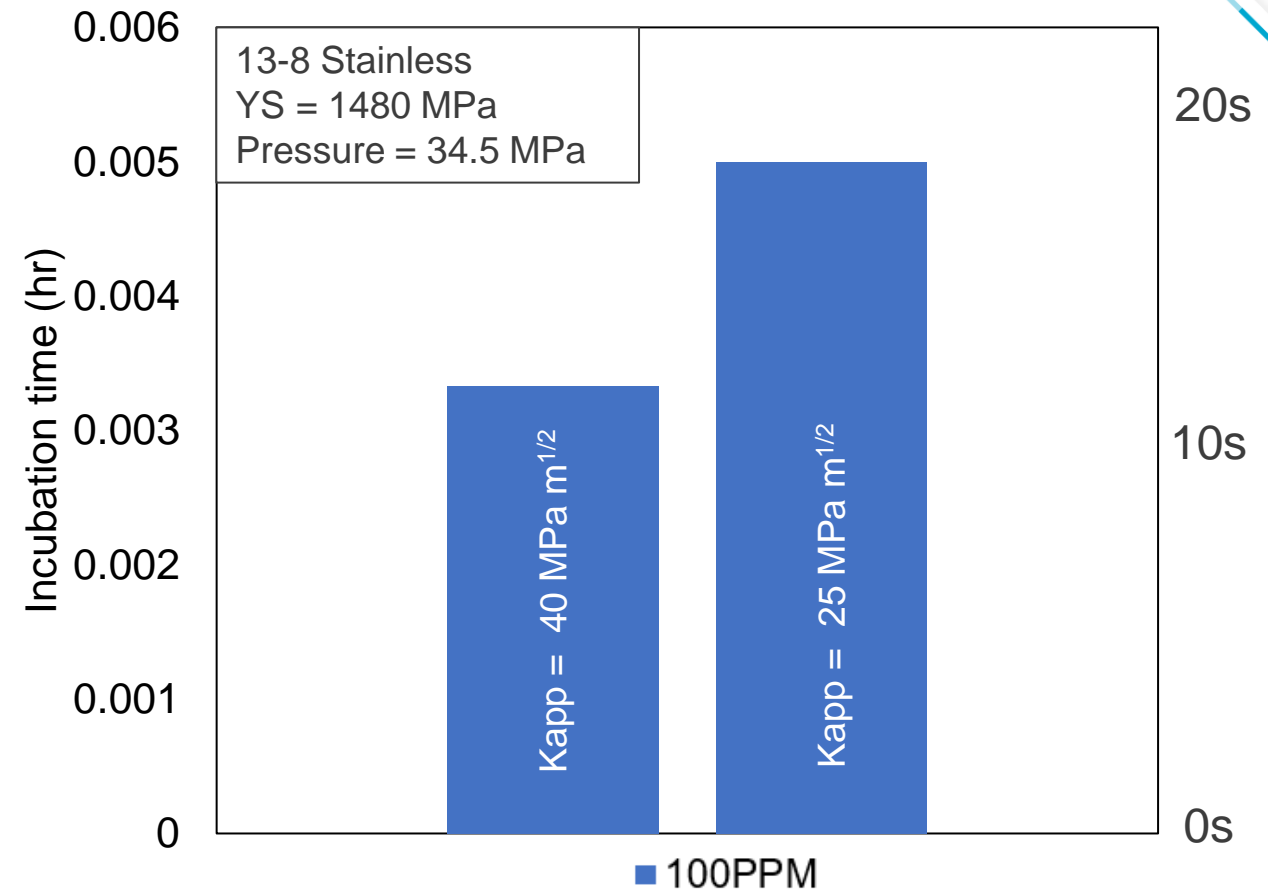
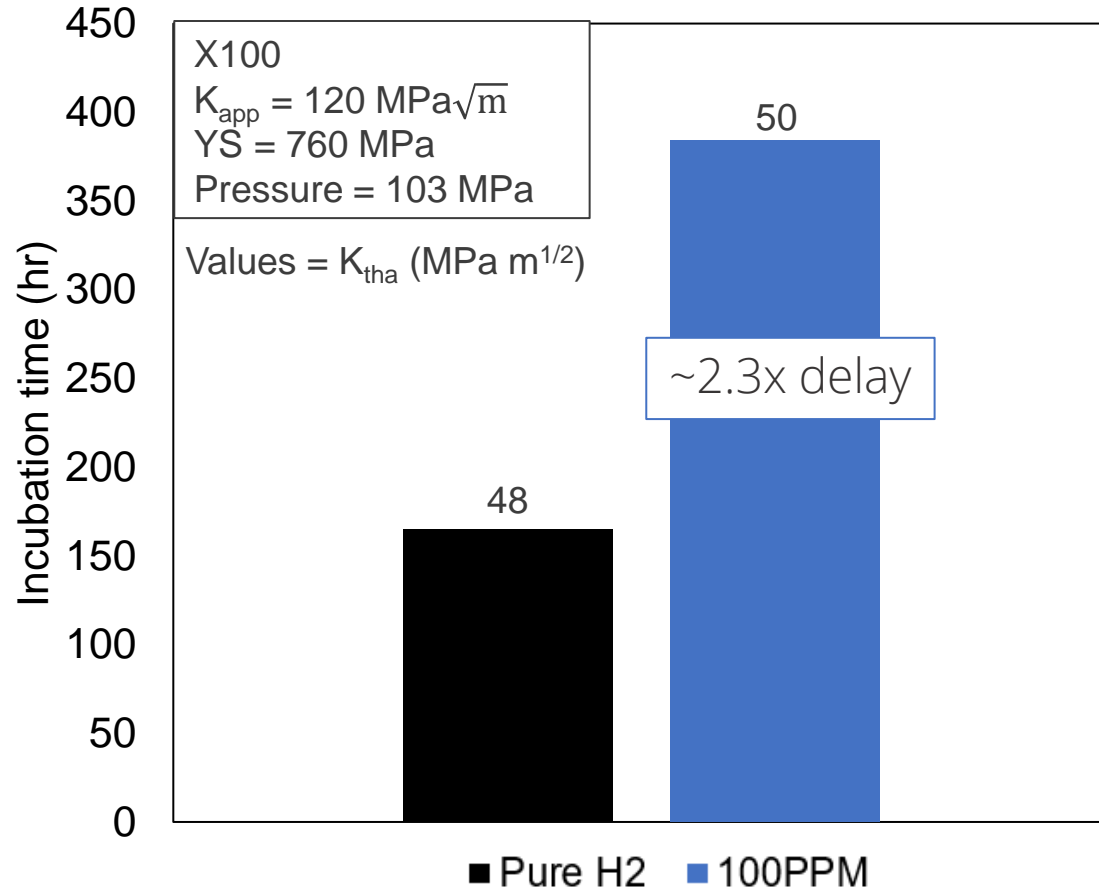
- Grade L shows a 5x delay at higher preload ($K_{app} = 60 \text{ MPa}\sqrt{\text{m}}$) and a 1.5x delay at lower preload ($K_{app} = 34 \text{ MPa}\sqrt{\text{m}}$)
- Similar crack arrest thresholds for all test conditions

100PPM & 1000PPM O₂ delay incubation time



- Similarly, the Grade J material showed delays of 15x at a higher preload ($K_{app} = 145 \text{ MPa}\sqrt{\text{m}}$) and a 2.2x delay increase at a lower preload ($K_{app} = 135 \text{ MPa}\sqrt{\text{m}}$)
- K thresholds were within $\pm 5 \text{ MPa}\sqrt{\text{m}}$ of average for both the pure and mixed gas conditions

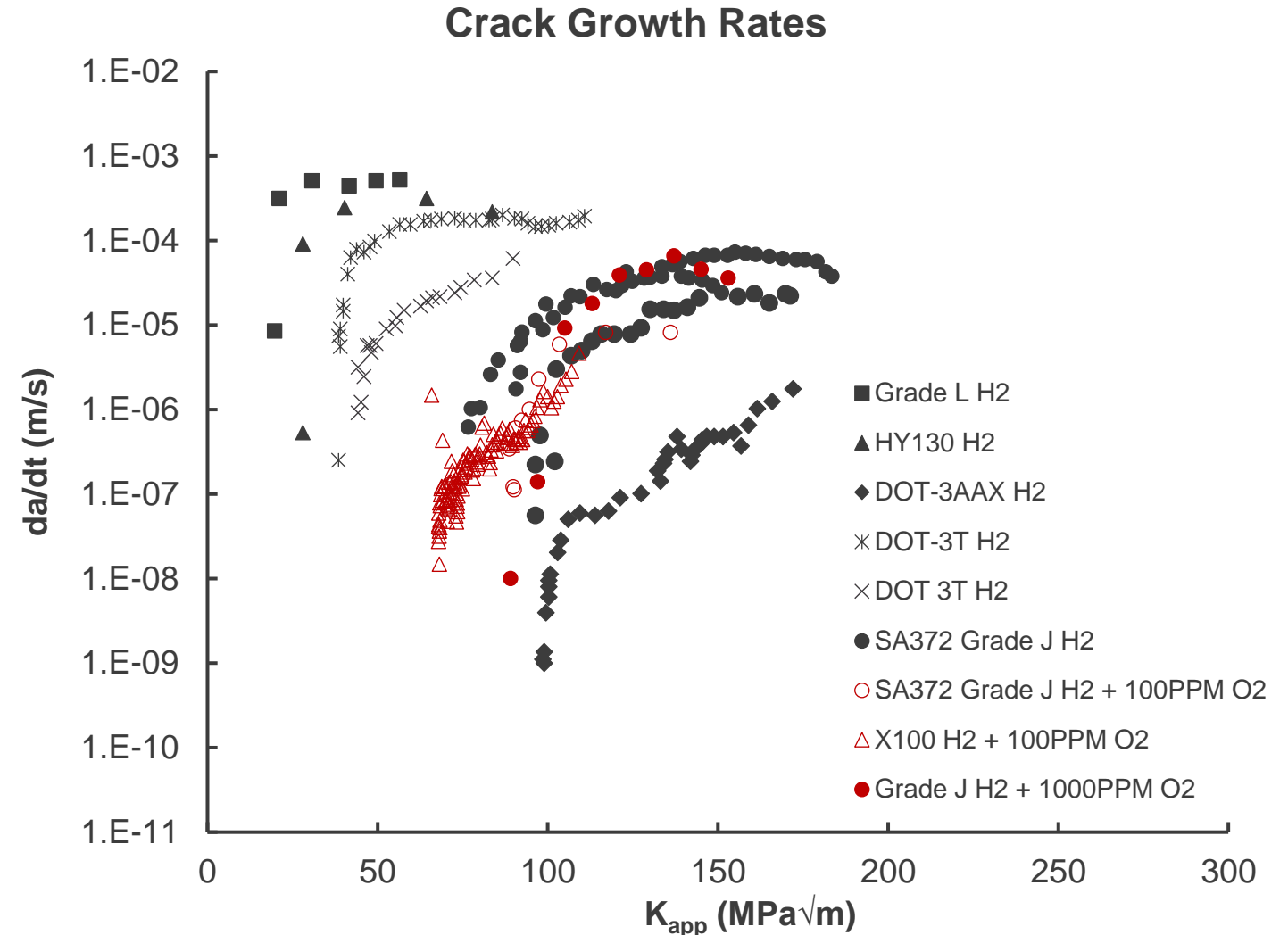
100PPM O₂ delays incubation time for X100, but 13-8 fractured immediately



- X100 also saw a delay with the addition of 100PPM O₂
- Both 13-8 samples fractured ($a/W > 97\%$) within seconds of exposure to H₂ + 100PPM O₂

Similar crack growth rates and arrest thresholds for both H₂ and mixed gas environment

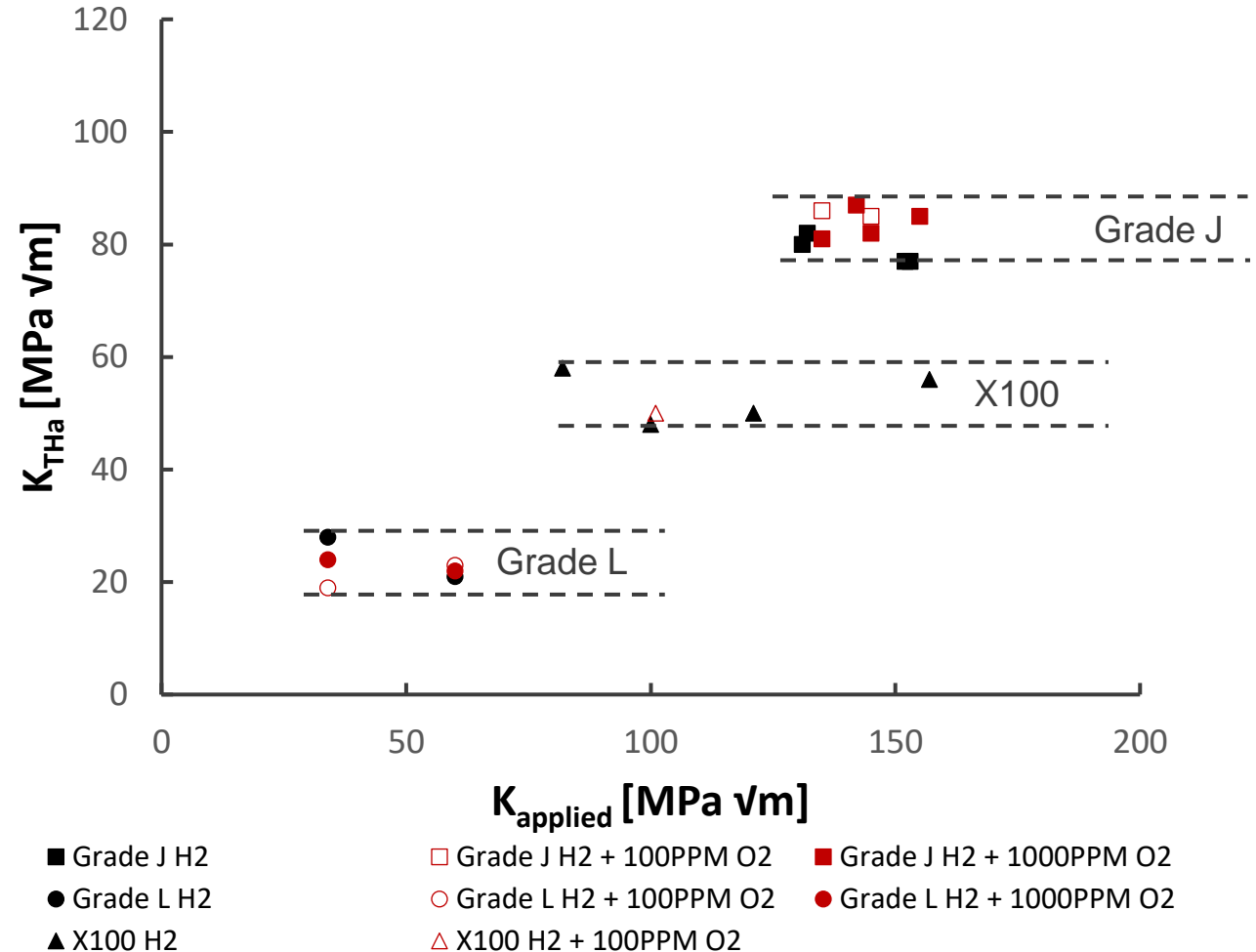
- Crack growth rates (da/dt) fall within the expected ranges from previous tests in pure hydrogen at similar pressures
- Adding oxygen to a high pressure hydrogen environment (103MPa) was shown to increase the time it took for subcritical cracks to propagate
 - The addition of oxygen impurities did not prevent crack propagation or effect crack arrest thresholds



Summary

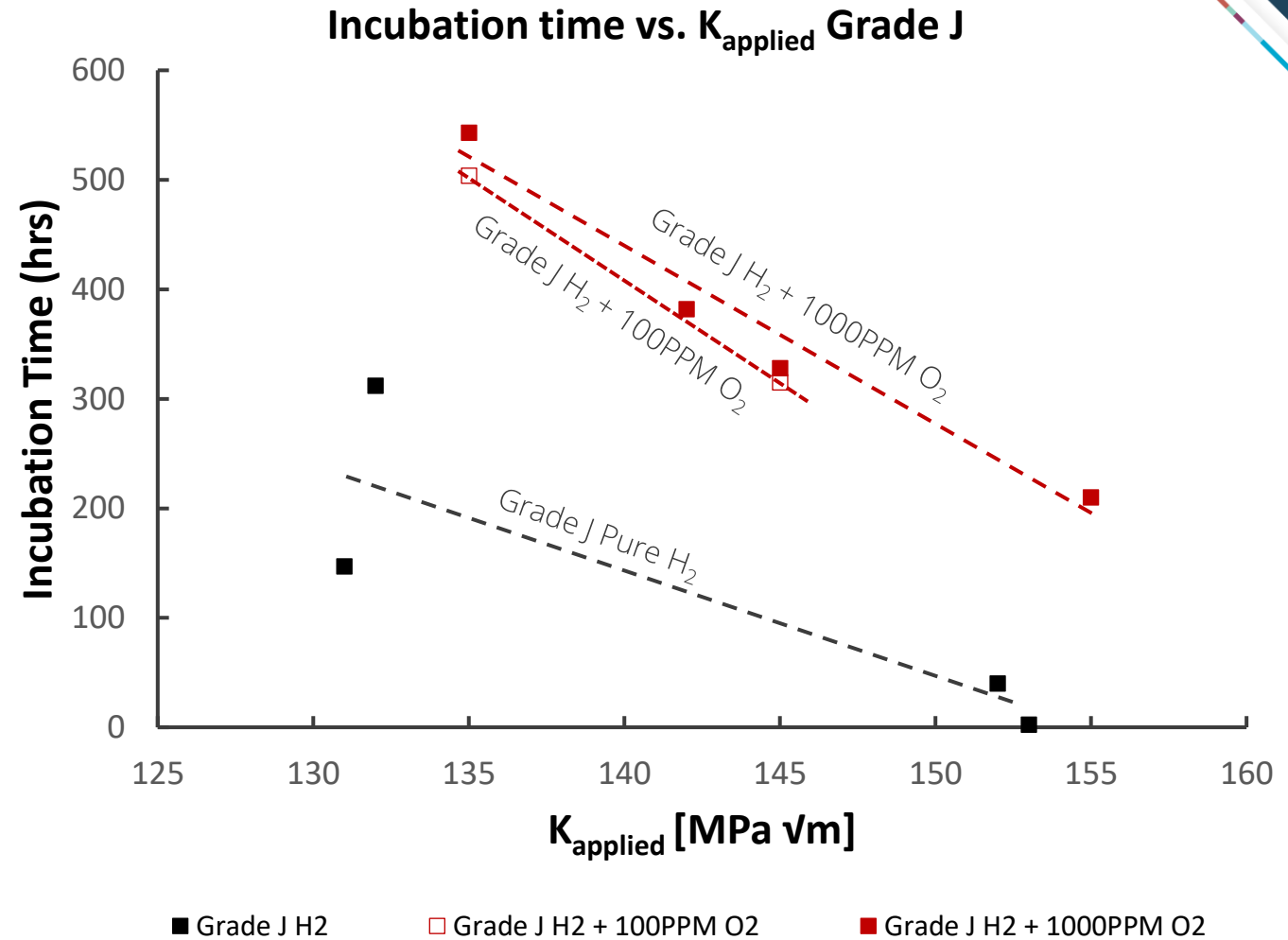
- The effects of low oxygen impurities in hydrogen gas on subcritical crack growth in high pressure hydrogen environment was studied
 - Constant displacement fracture tests were carried out in pure hydrogen and mixed gas (100 and 1000PPM oxygen) environments at 103MPa (15ksi)
- K_{THa} appears to be independent from oxygen content
 - All tests with pure hydrogen and oxygen impurities fall within an apx. $10\text{MPa}\sqrt{\text{m}}$ range
- The 13-8 material fractured nearly through the width of the sample during initial pressurization

Crack arrest threshold vs. applied preload



Summary

- Introducing 100PPM oxygen increased the incubation time by factors between 1.5x and 15x, but did not prevent crack propagation
- Increasing the oxygen content from 100PPM to 1000PPM further delayed the incubation time, but had a smaller relative effect than going from pure hydrogen to hydrogen + 100PPM oxygen
 - The relative increase in incubation time when moving from 100 to 1000PPM oxygen was minor for the Grade J material
- Based on this data, low oxygen impurities should not be relied upon for long-term mitigation of hydrogen embrittlement
- We would like to acknowledge James McNair, Jeff Campbell, and Brendan Davis for their assistance with the experimental setup



**Thank you for your
attention!**

Rob Wheeler
rwheel@sandia.gov

