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# Subcritical Crack Growth in High-pressure Hydrogen and Hydrogen With Oxygen Impurity

Robert W. Wheeler, Chris San Marchi, and Joseph Ronevich

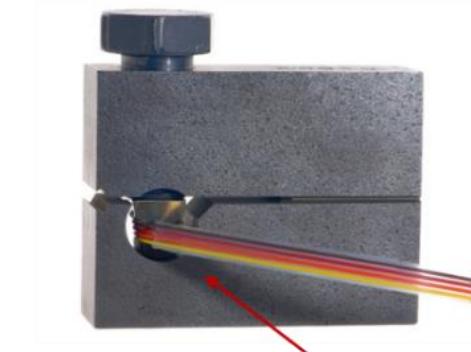
Hydrogen and Materials Science Department, Sandia National Laboratories Livermore, CA, USA

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# Overview

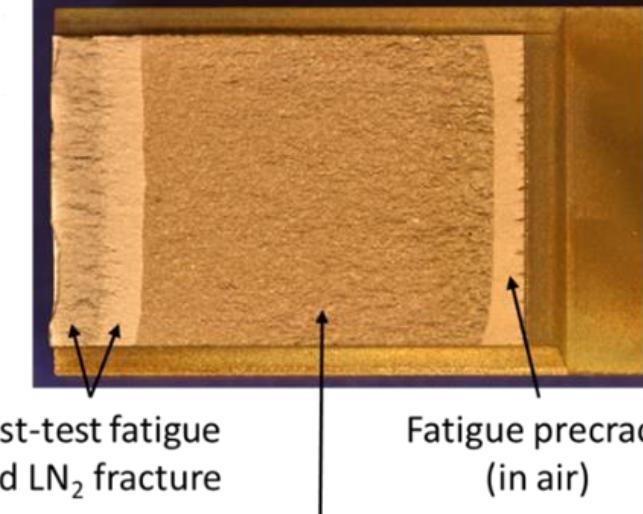
- Background and Motivation
- Experimental Methods
- Subcritical Crack Growth in High-pressure Hydrogen and Hydrogen With Oxygen Impurity
  - SA372 Grade J steel
  - SA372 Grade L steel
  - X100 pipeline steel
  - 13-8 stainless steel
- Summary and Conclusions

Wedge-opened Loaded (WOL) Sample



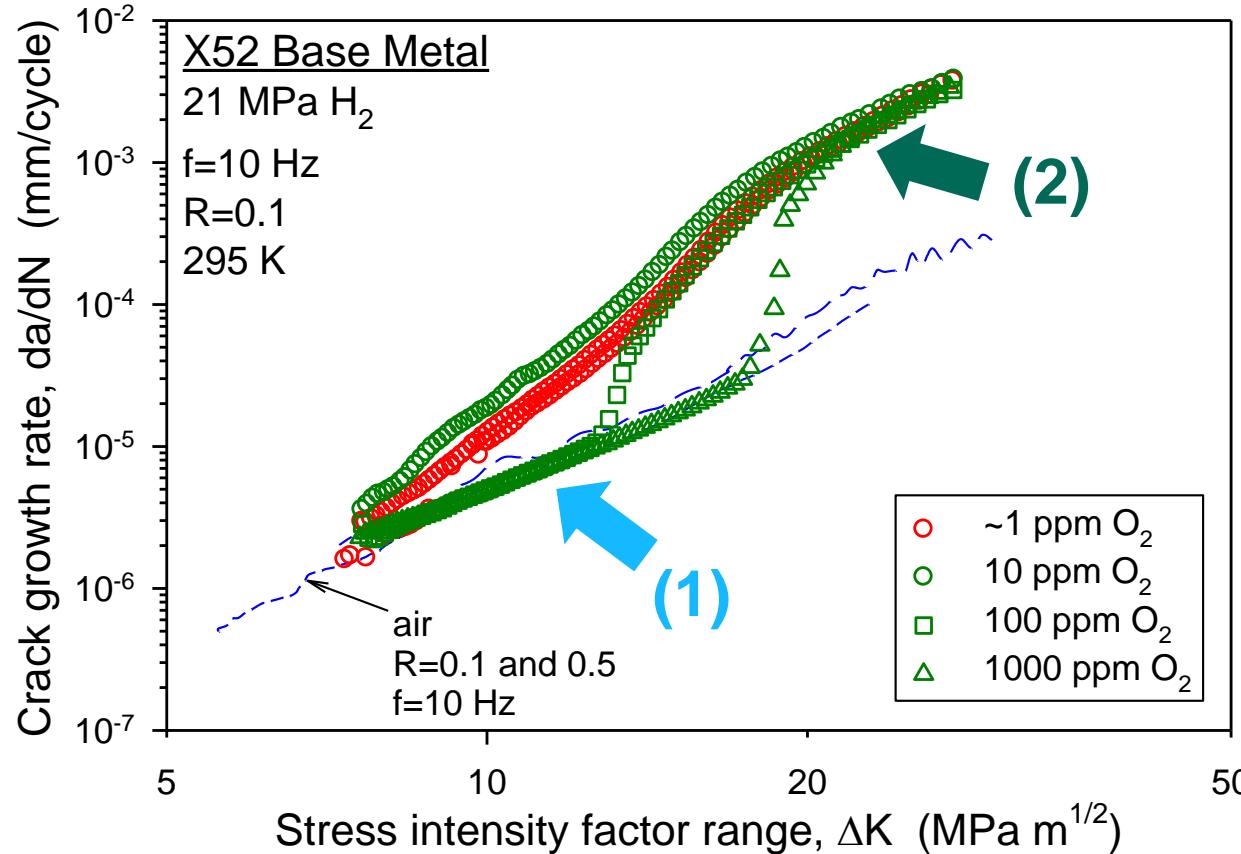
Reaction pin (load tup) with externally monitored strain gauge

Grade L: WOL Fracture Surface



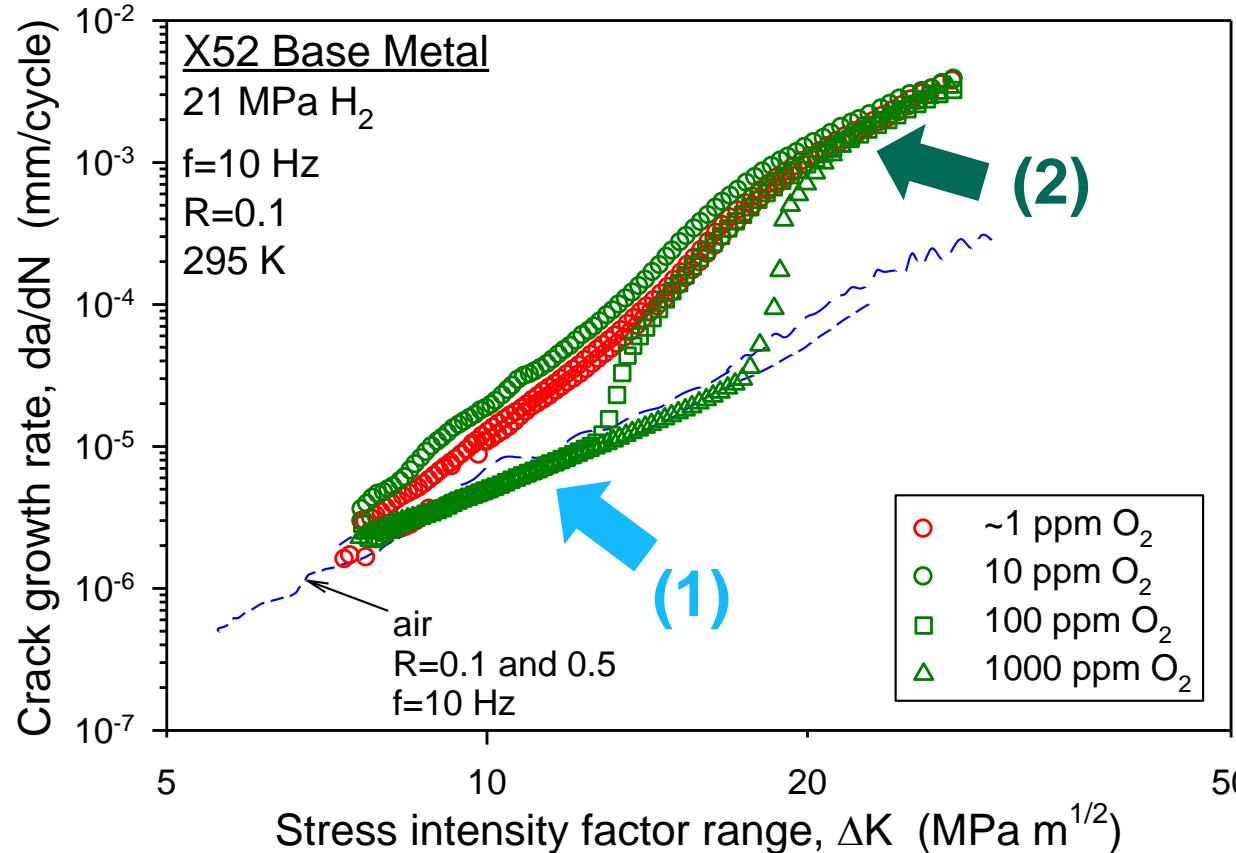
Crack growth in  $H_2 +$   
100PPM  $O_2$

# Oxygen is known to affect measurements of fatigue and fracture



- Numerous examples of trace gases mitigating fatigue crack growth rate (FCGR) in laboratory conditions
- Example:
  - (1) Oxygen reduces FCGR to air
  - (2) Oxygen has no effect on FCGR in  $H_2$

# Oxygen is known to affect measurements of fatigue and fracture

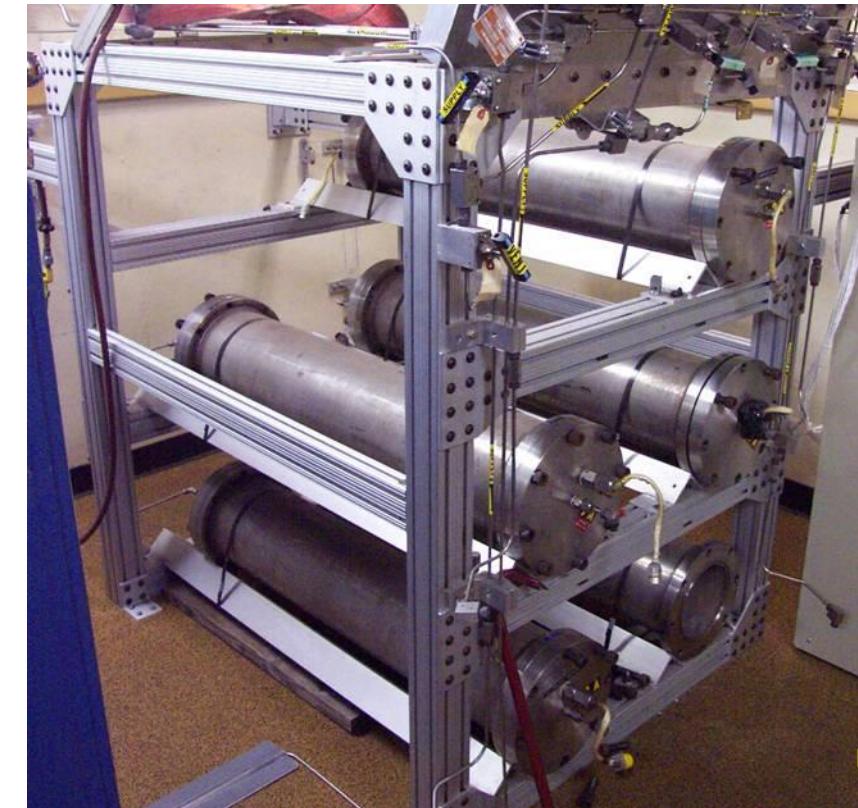
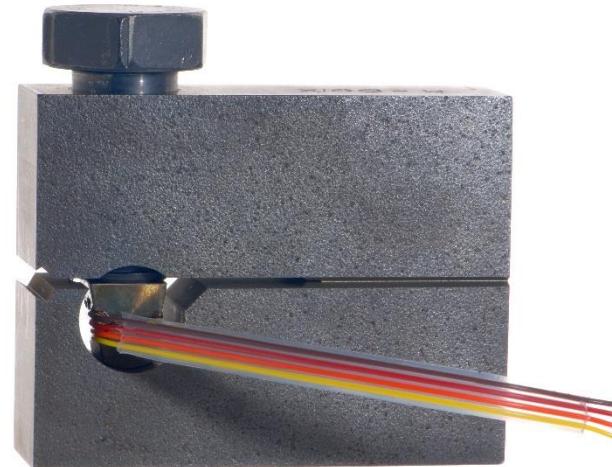


- Fatigue crack growth tests are typically performed at 1 Hz ( $\pm$ decade)
  - $da/dN = 10^{-5}$  mm/cycle
  - Time for  $\Delta a = 1$  mm:  $\sim 1$  day
  - 1 day = 0.02% of 10 year life
- Are the time scales of a typical laboratory fatigue test sufficient to demonstrate kinetics over decades?
  - More accurately simulate the mechanical/environmental conditions that components see when in use
  - Does trace oxygen have long term mitigation effects on hydrogen embrittlement?

# Sustained load testing can be executed over periods of days to weeks to months to years

- Fixed displacement tests
- Placed in pressure vessels & pressurized up to 140 MPa gaseous environment
  - Experiments in this study were performed at 103 MPa
- Test durations can range from days to years
- Instrumented reaction pins allows us to determine incubation time
- Directly compare subcritical crack growth in hydrogen and mixed gas environments

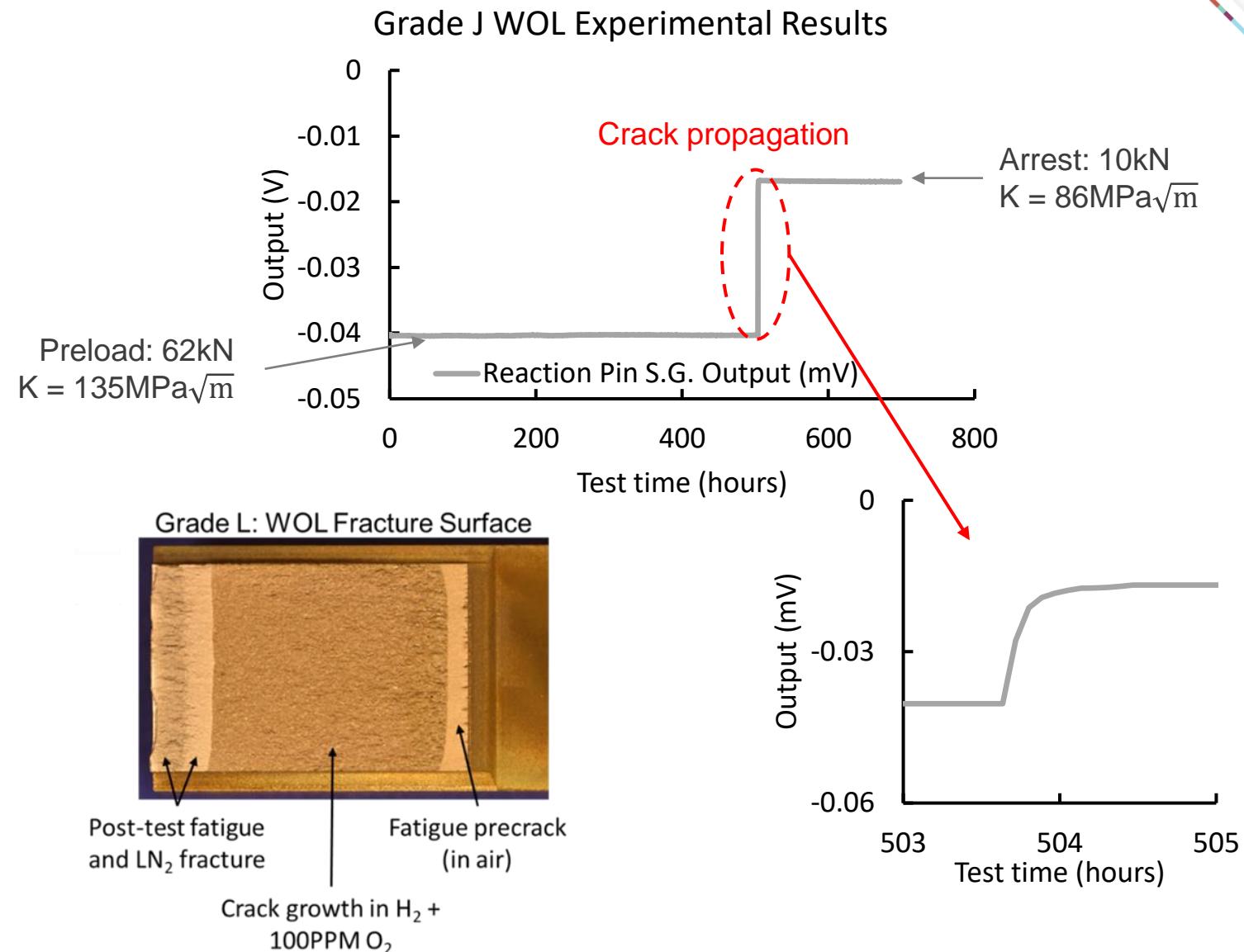
**Wedge-opened loaded (WOL)**



**ASTM E1681 – Threshold Stress Intensity Factor for Environment-Assisted Cracking**

# Crack initiation and growth rates can be measured during constant displacement fracture experiments

- Instrumented reaction pins allow for determination of incubation time and crack growth rates
  - Continuous data collection throughout the duration of the experiments
- Time between the initial crack propagation and arrest can range between seconds to hours
  - With a constant displacement, the crack growth rates can be determined from the load on the reaction pin
- Post-test fatigue and heat tinting are used to mark fracture surfaces



# Material selection and fracture surfaces

- SA372 Grade J steel
  - YS = 700 MPa
- SA372 Grade L steel
  - YS = 730 MPa
- X100 pipeline steel
  - YS = 760 MPa
- Precipitation Hardened 13-8 stainless steel
  - YS = 1480 MPa

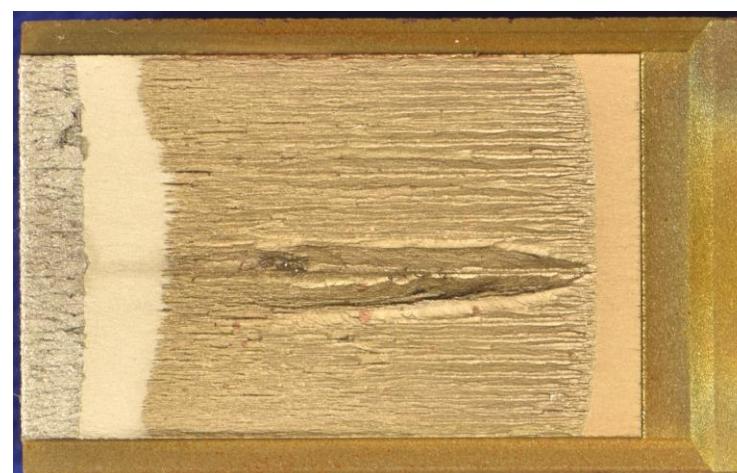
Grade L



Grade J



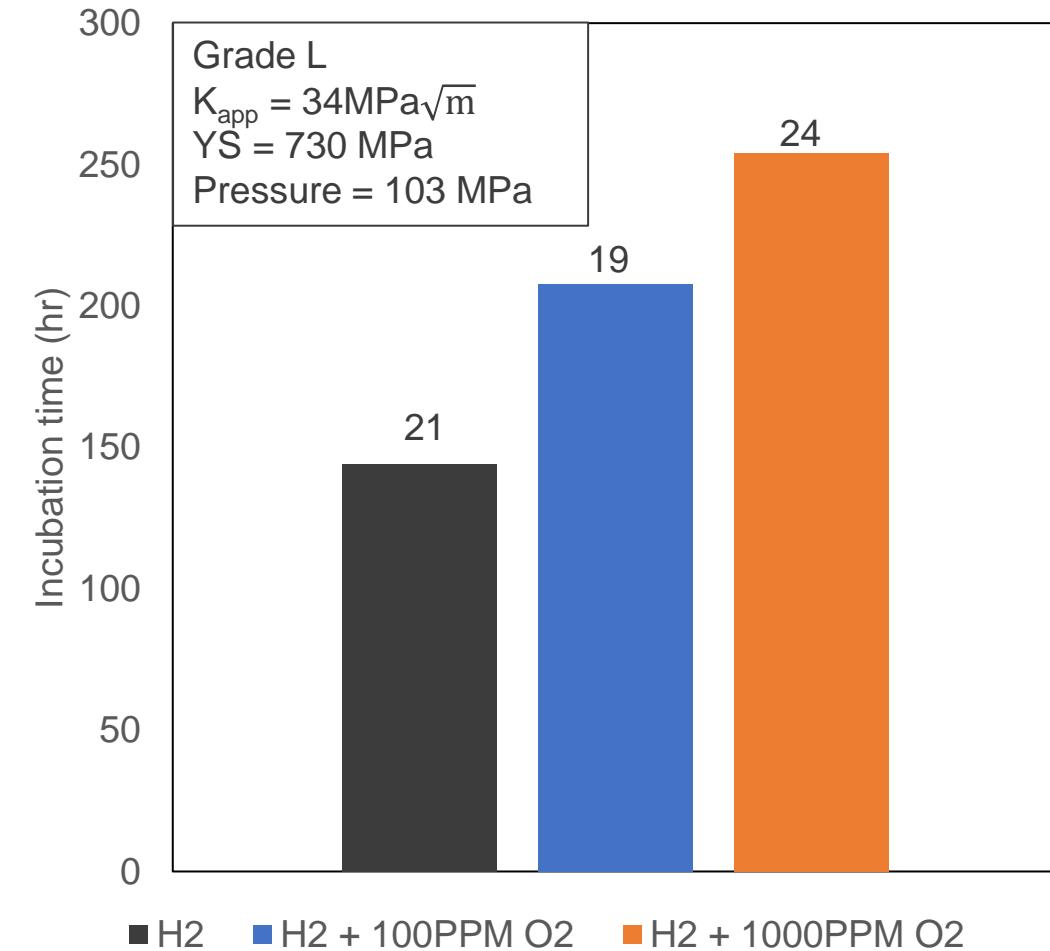
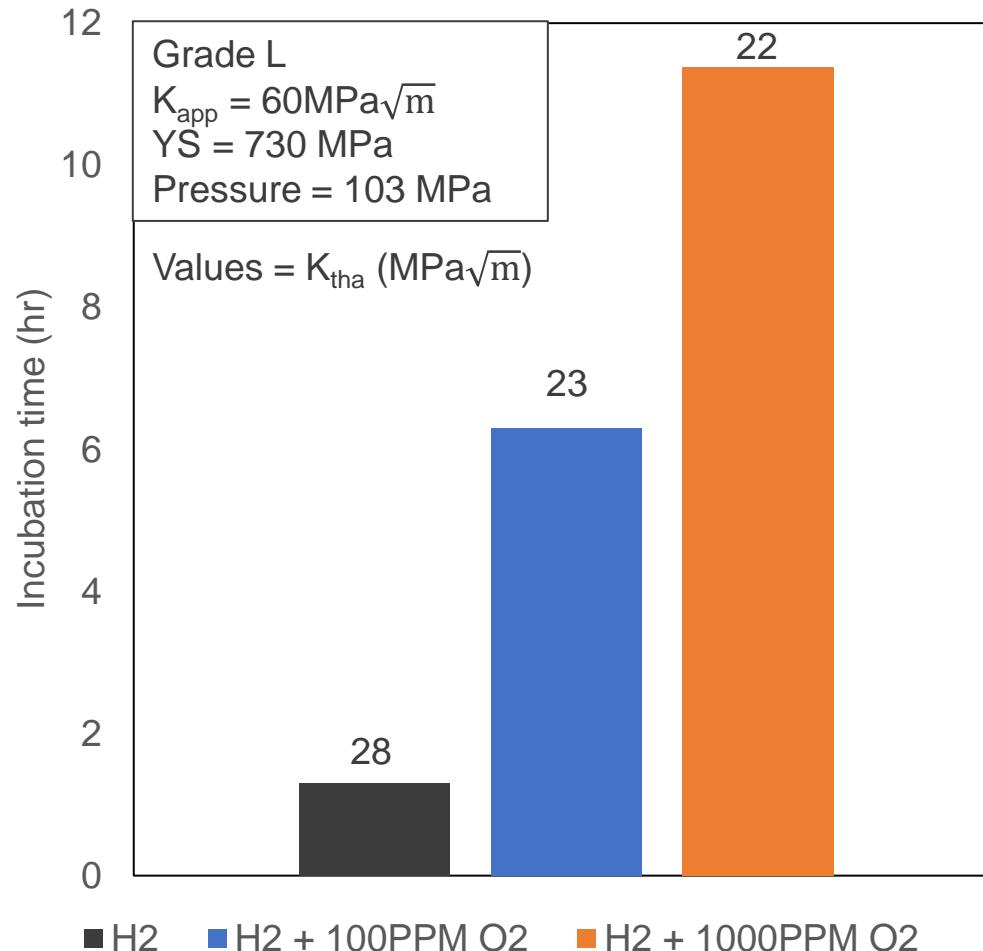
X100



13-8

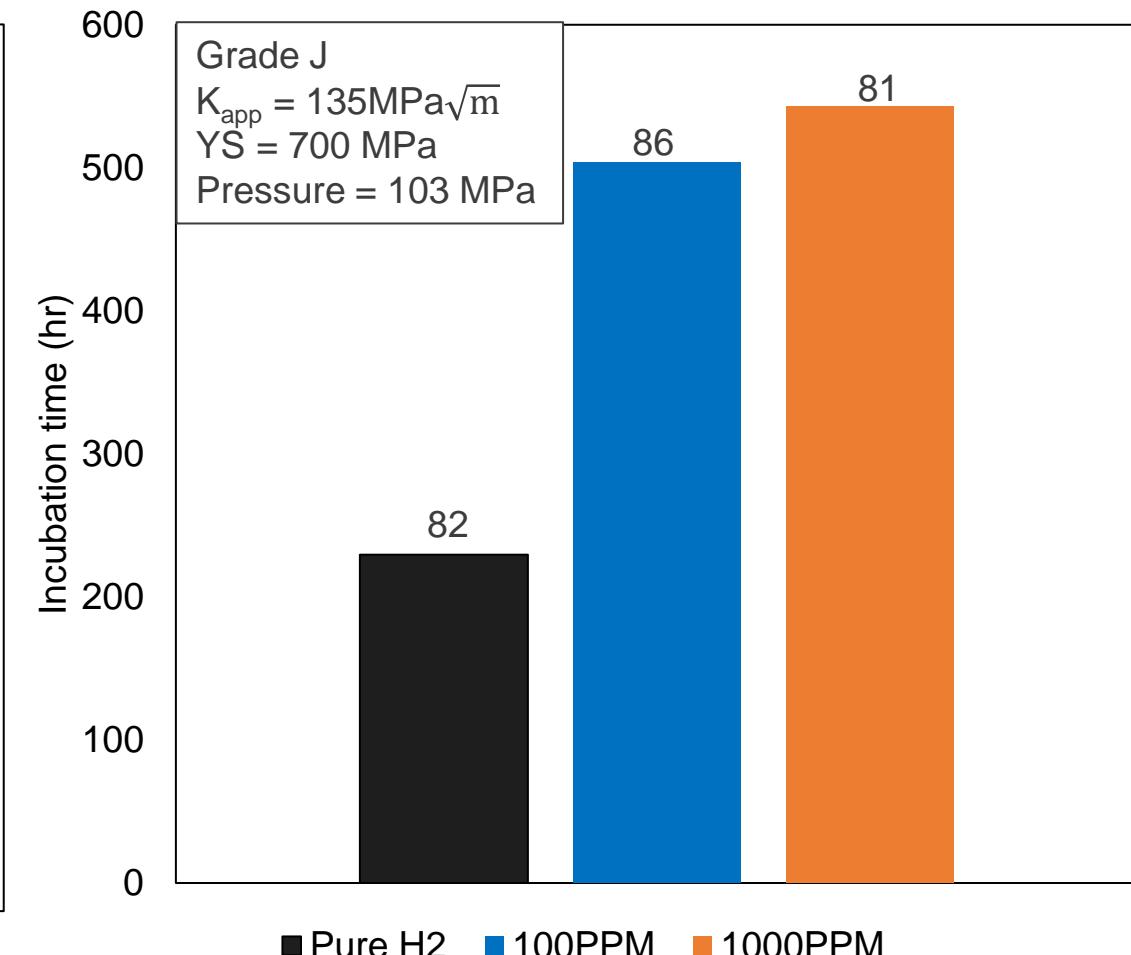
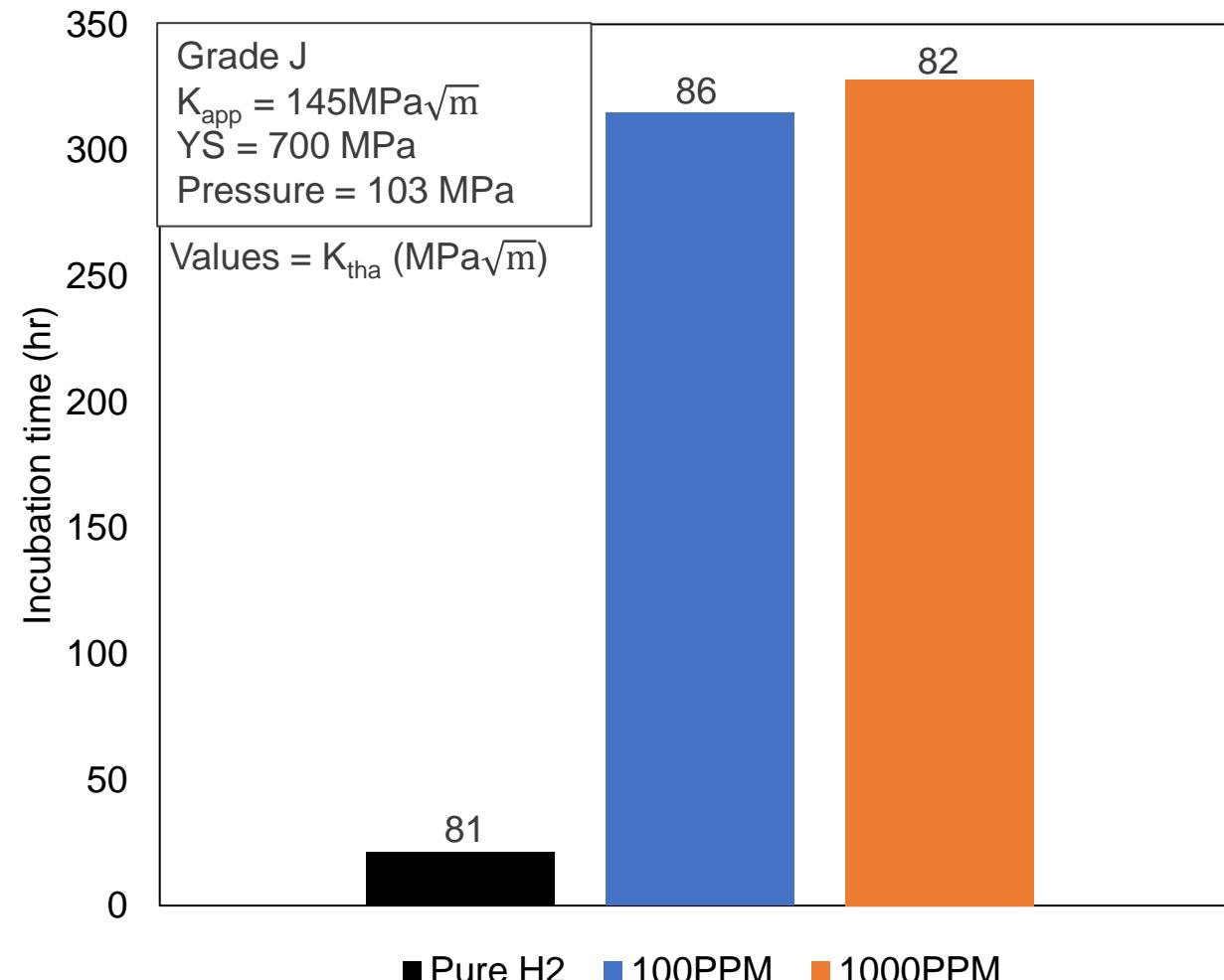


# 100PPM & 1000PPM O<sub>2</sub> delay incubation time



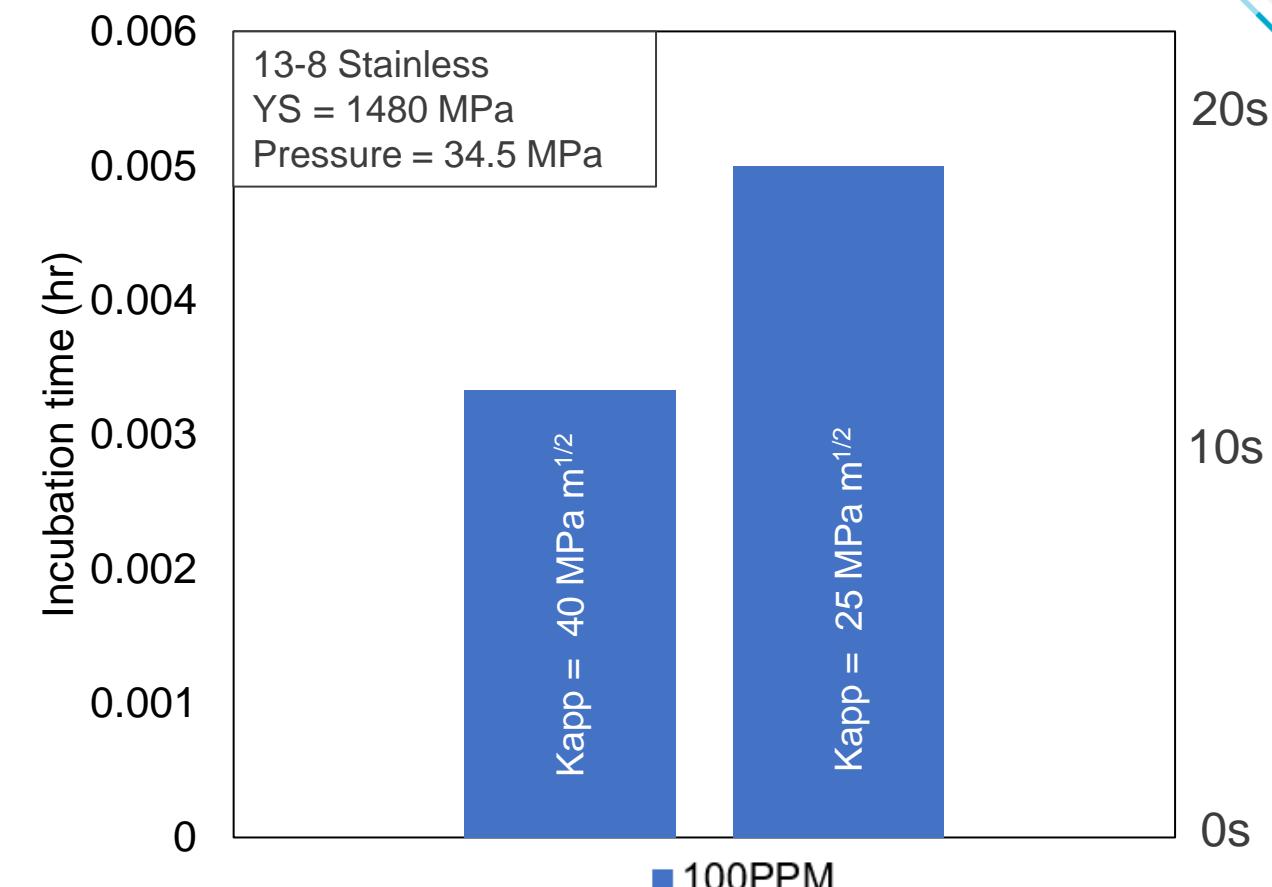
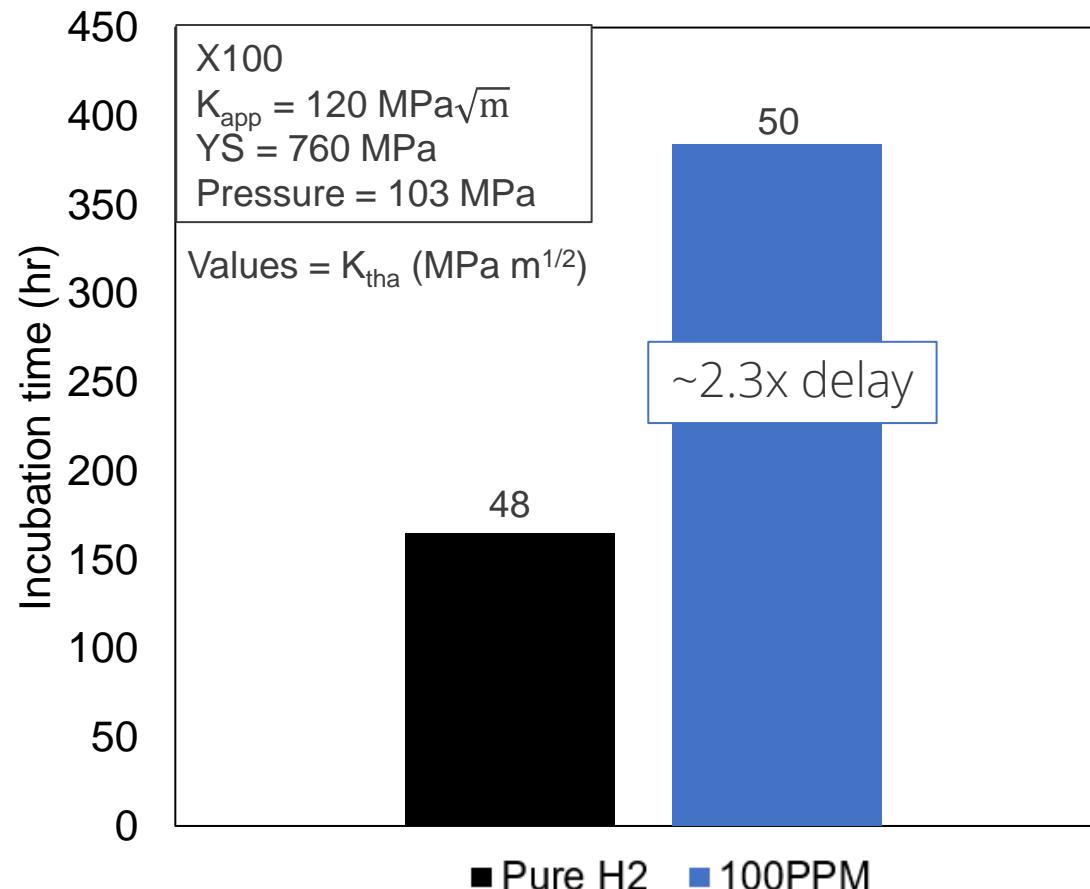
- Grade L shows a 5x delay at higher preload ( $K_{app} = 60 \text{ MPa}\sqrt{\text{m}}$ ) and a 1.5x delay at lower preload ( $K_{app} = 34 \text{ MPa}\sqrt{\text{m}}$ )
- Similar crack arrest thresholds for all test conditions

# 100PPM & 1000PPM O<sub>2</sub> delay incubation time



- Similarly, the Grade J material showed delays of 15x at a higher preload ( $K_{app} = 145 \text{ MPa}\sqrt{\text{m}}$ ) and a 2.2x delay increase at a lower preload ( $K_{app} = 135 \text{ MPa}\sqrt{\text{m}}$ )
- $K$  thresholds were within  $\pm 5 \text{ MPa}\sqrt{\text{m}}$  of average for both the pure and mixed gas conditions

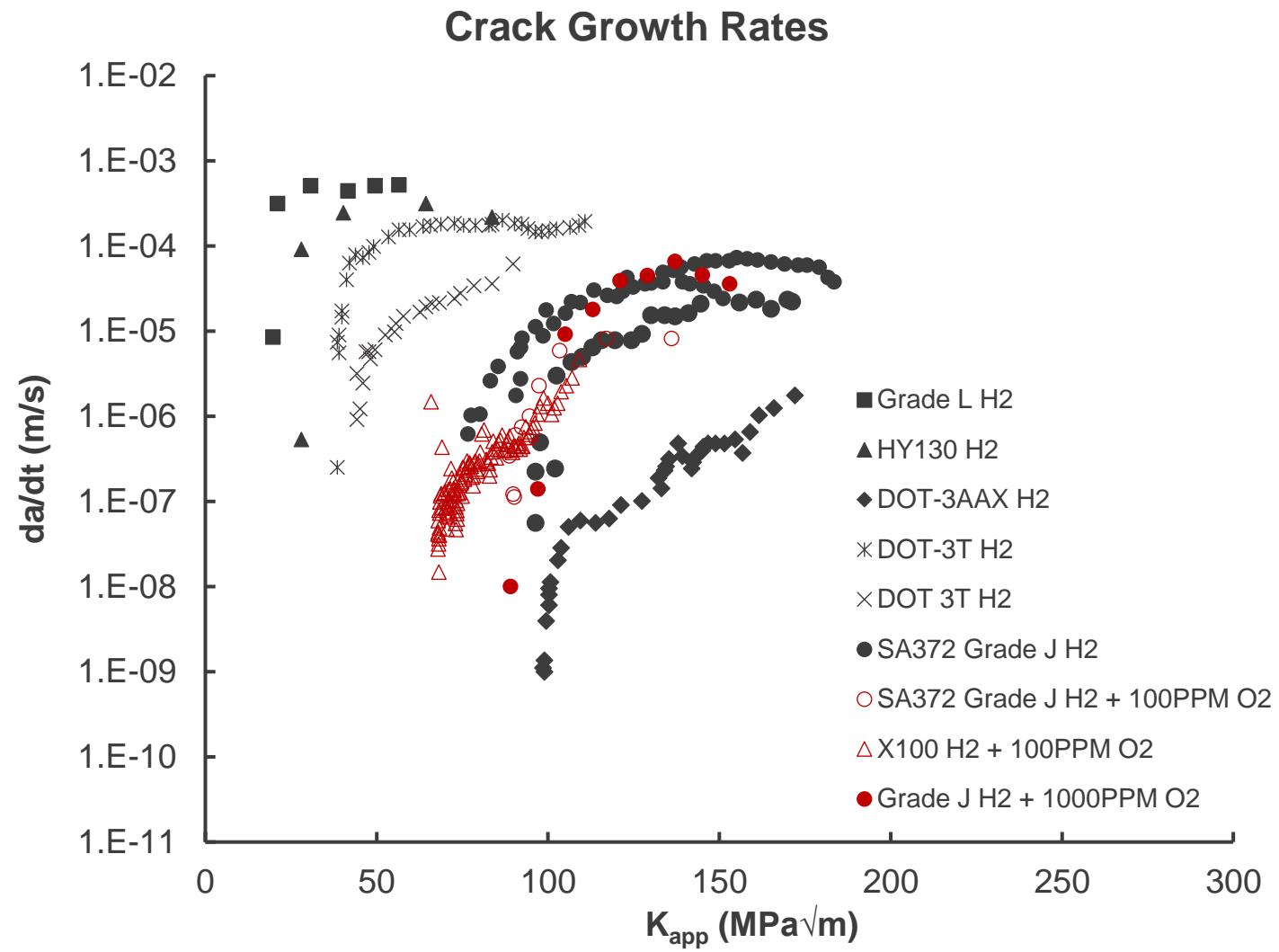
# 100PPM O<sub>2</sub> delays incubation time for X100, but 13-8 fractured immediately



- X100 also saw a delay with the addition of 100PPM O<sub>2</sub>
- Both 13-8 samples fractured ( $a/W > 97\%$ ) within seconds of exposure to H<sub>2</sub> + 100PPM O<sub>2</sub>

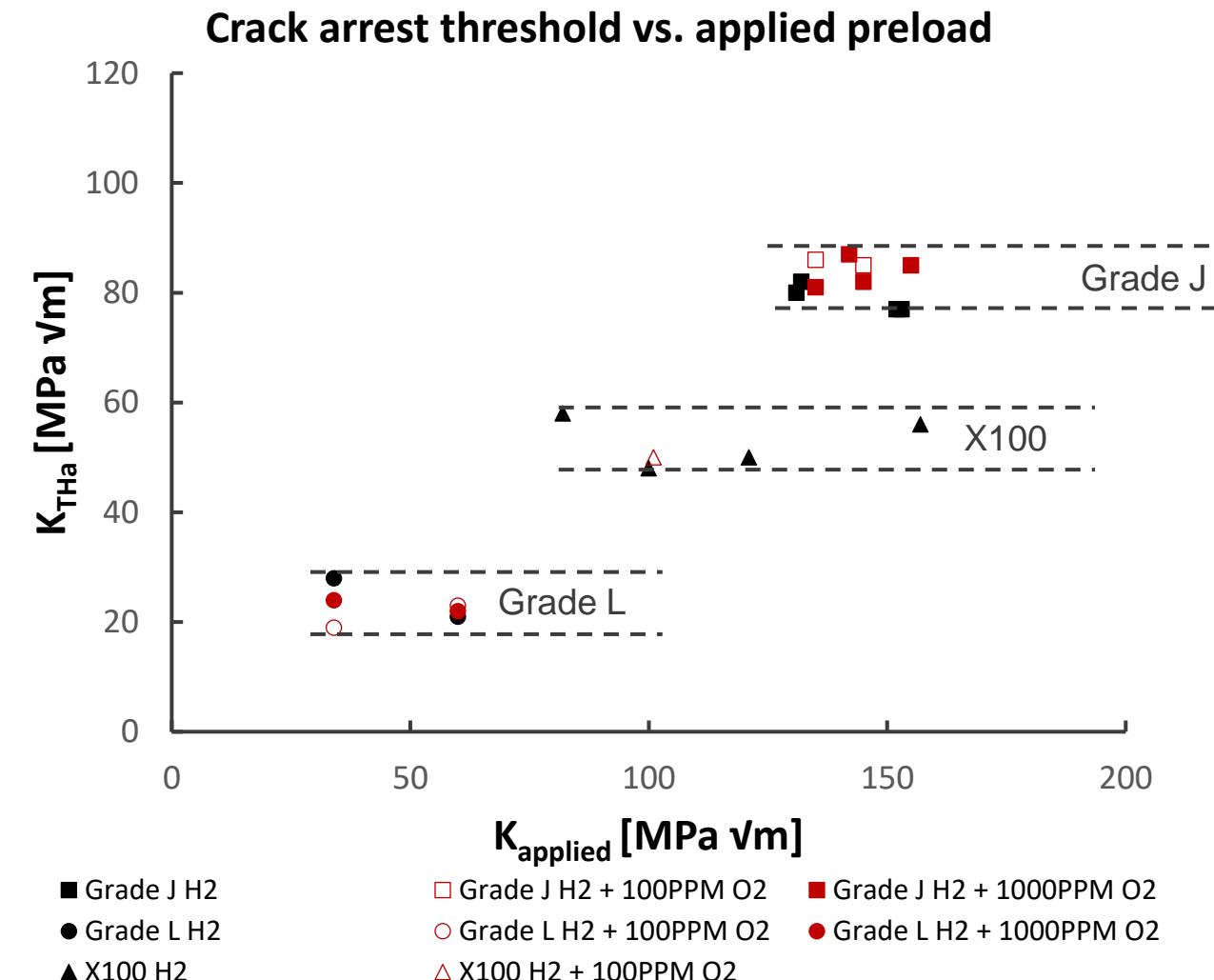
# Similar crack growth rates and arrest thresholds for both H<sub>2</sub> and mixed gas environment

- Crack growth rates ( $da/dt$ ) fall within the expected ranges from previous tests in pure hydrogen at similar pressures
- Adding oxygen to a high pressure hydrogen environment (103MPa) was shown to increase the time it took for subcritical cracks to propagate
  - The addition of oxygen impurities did not prevent crack propagation or effect crack arrest thresholds



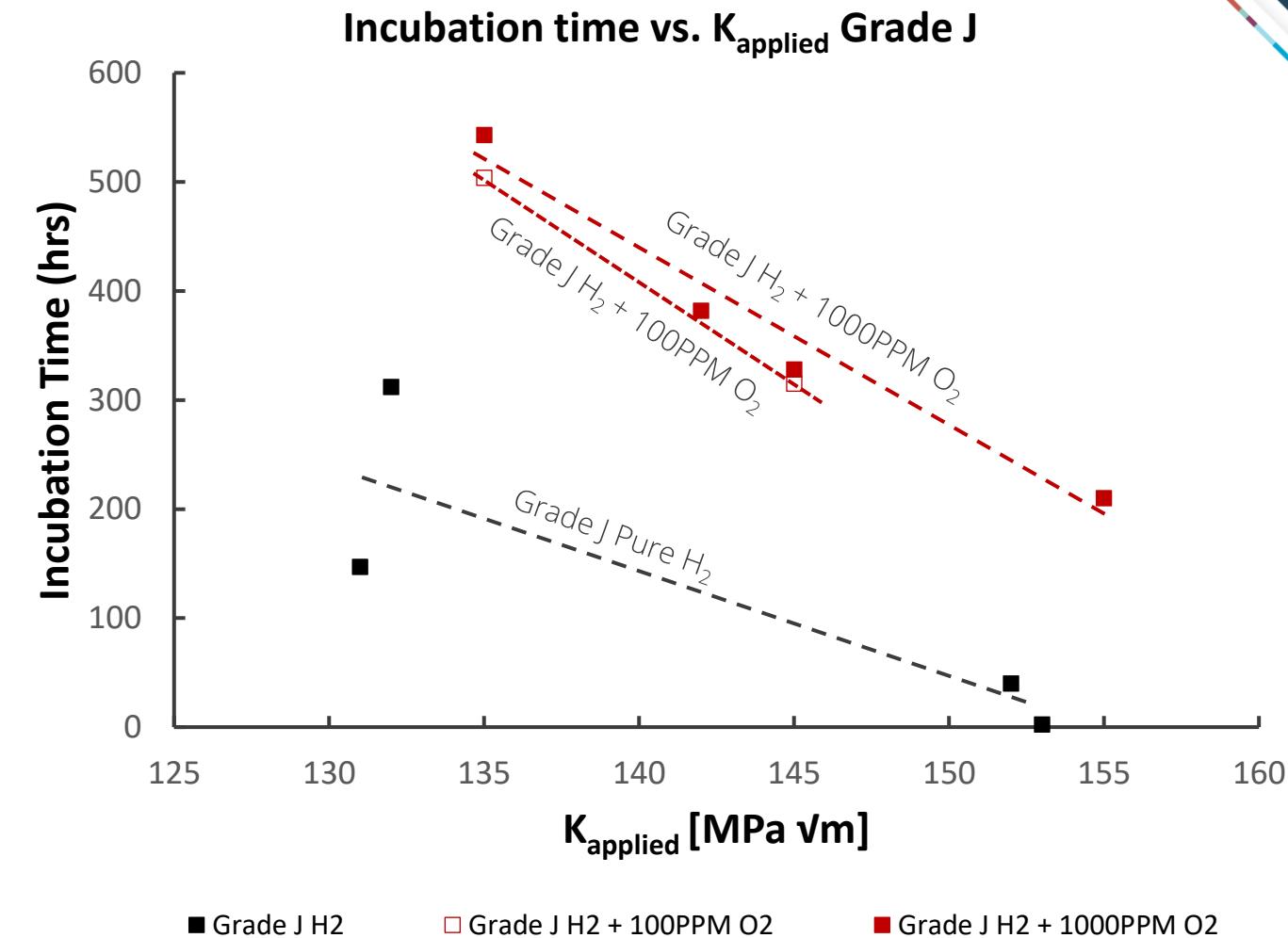
# Summary

- The effects of low oxygen impurities in hydrogen gas on subcritical crack growth in high pressure hydrogen environment was studied
  - Constant displacement fracture tests were carried out in pure hydrogen and mixed gas (100 and 1000PPM oxygen) environments at 103MPa (15ksi)
- $K_{THa}$  appears to be independent from oxygen content
  - All tests with pure hydrogen and oxygen impurities fall within an apx.  $10\text{ MPa}\sqrt{\text{m}}$  range
- The 13-8 material fractured nearly through the width of the sample during initial pressurization



# Summary

- Introducing 100PPM oxygen increased the incubation time by factors between 1.5x and 15x, but did not prevent crack propagation
- Increasing the oxygen content from 100PPM to 1000PPM further delayed the incubation time, but had a smaller relative effect than going from pure hydrogen to hydrogen + 100PPM oxygen
  - The relative increase in incubation time when moving from 100 to 1000PPM oxygen was minor for the Grade J material
- Based on this data, low oxygen impurities should not be relied upon for long-term mitigation of hydrogen embrittlement
- We would like to acknowledge James McNair, Jeff Campbell, and Brendan Davis for their assistance with the experimental setup



Thank you for your  
attention!

Rob Wheeler  
[rwheel@sandia.gov](mailto:rwheel@sandia.gov)