



Exceptional service in the national interest

Exploring the behavior of MoNbTaTi refractory CCAs across composition space using a machine-learned interatomic potential

Megan McCarthy^a, Jacob Startt^b, Remi Dingreville^b, Aidan Thompson^a, Mitchell Wood^a

^a Center for Computing Research, Sandia National Laboratories – New Mexico

^b Center for Integrated Nanotechnologies, Sandia National Laboratories – New Mexico

Multiscale Materials Modeling (MMM10)

October 6th, 2022



Sandia National Laboratories is a multission laboratory managed and operated by National Technology and Engineering Solutions of Sandia LLC, a wholly owned subsidiary of Honeywell International Inc. for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

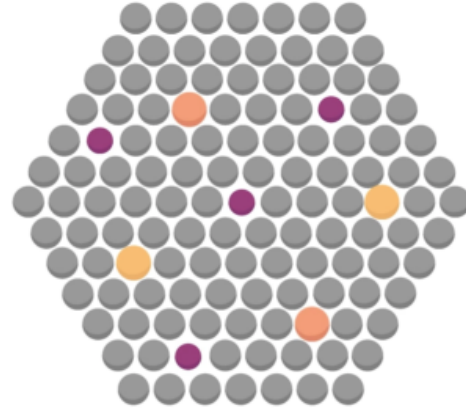
Sandia National Laboratories is a multission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.



Complex concentrated alloys



<https://doi.org/10.21203/rs.2.15081/v1>



Conventional alloy



High-entropy alloy

CCA: Alloy with high concentrations of 3 or more elements that can coexist without extreme phase separation

AKA: high-entropy alloy (HEA), multi-principal element alloy (MPEA)

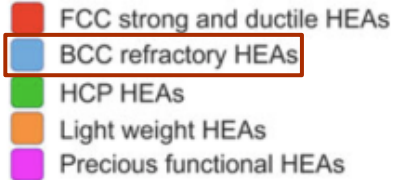
3



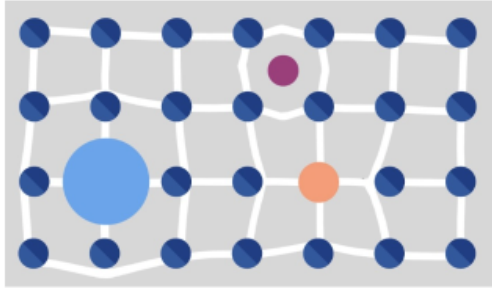
<https://doi.org/10.21203/rs.2.15081/v1>



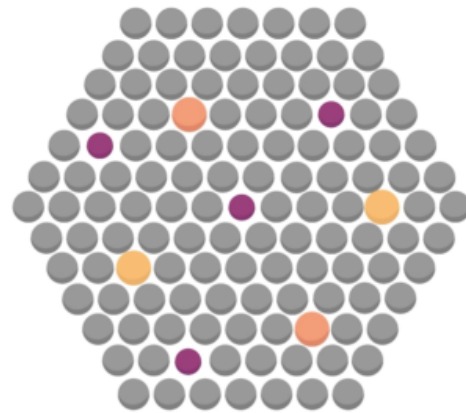
AKA: high-entropy alloy (HEA), multi-principal element alloy (MPEA)



Complex concentrated alloys



<https://doi.org/10.21203/rs.2.15081/v1>



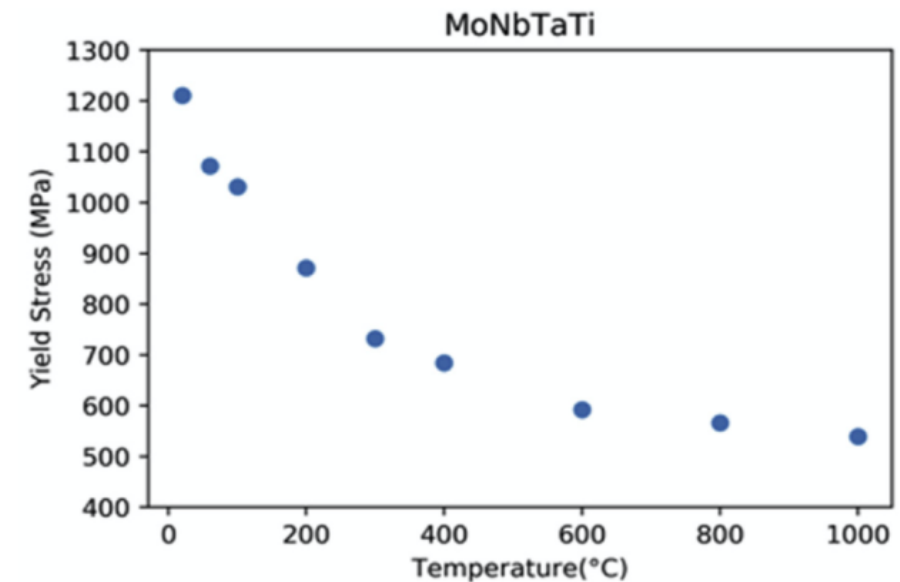
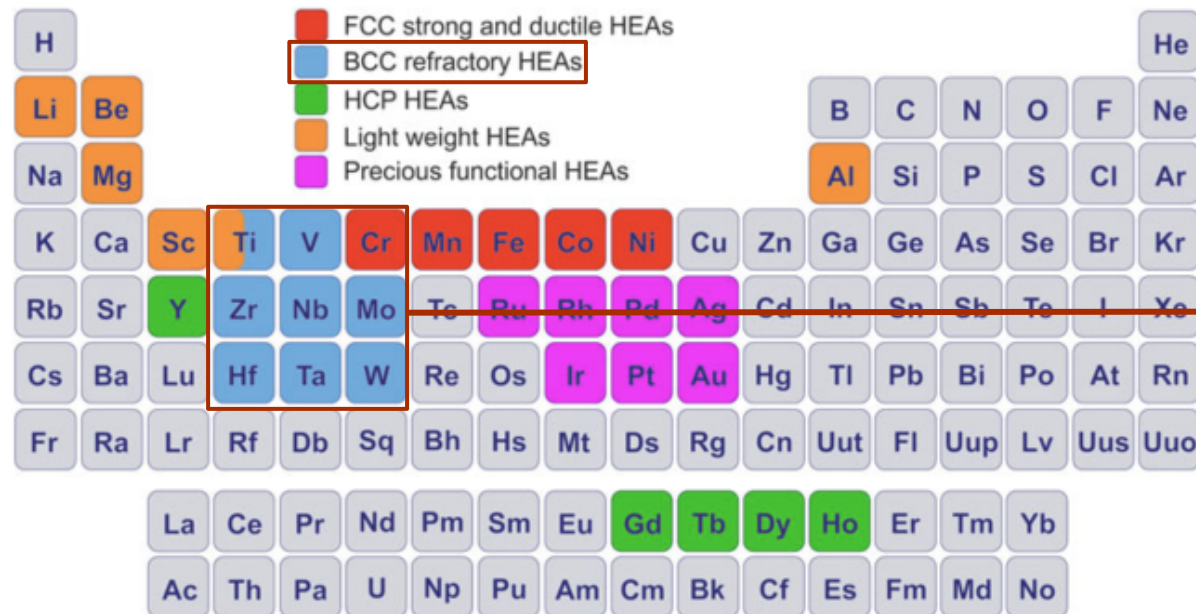
Conventional alloy



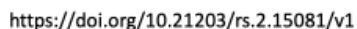
High-entropy alloy

CCA: Alloy with high concentrations of 3 or more elements that can coexist without extreme phase separation

AKA: high-entropy alloy (HEA), multi-principal element alloy (MPEA)

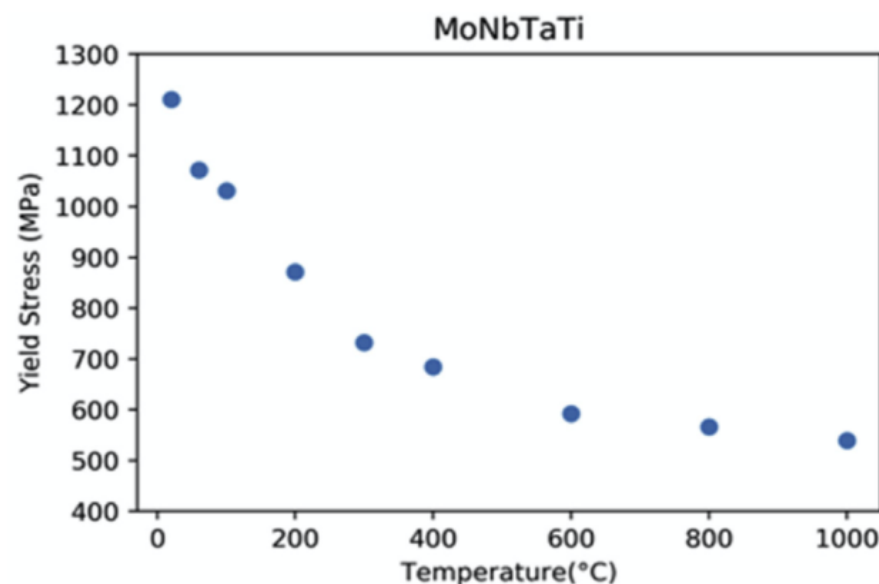
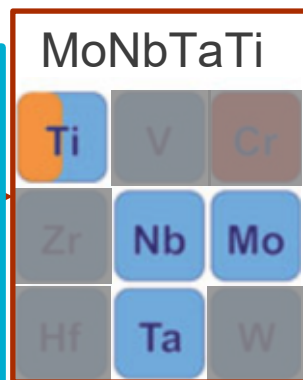
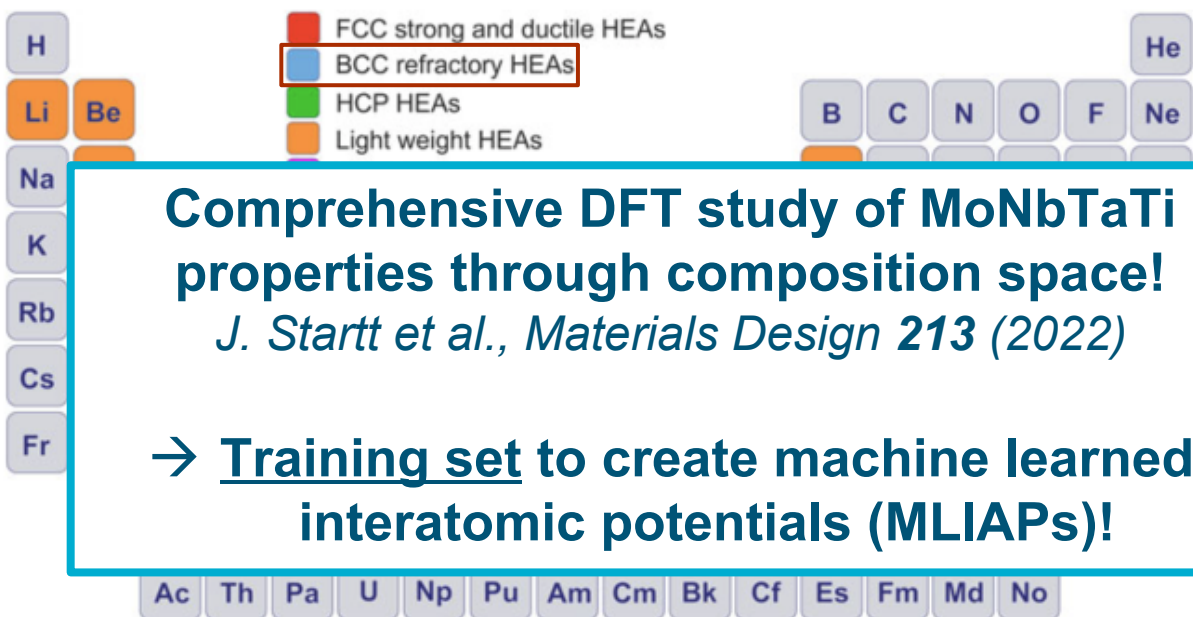


Coury et al., Acta Mat. 175 (2019)



CCA: Alloy with high concentrations of 3 or more elements that can coexist without extreme phase separation

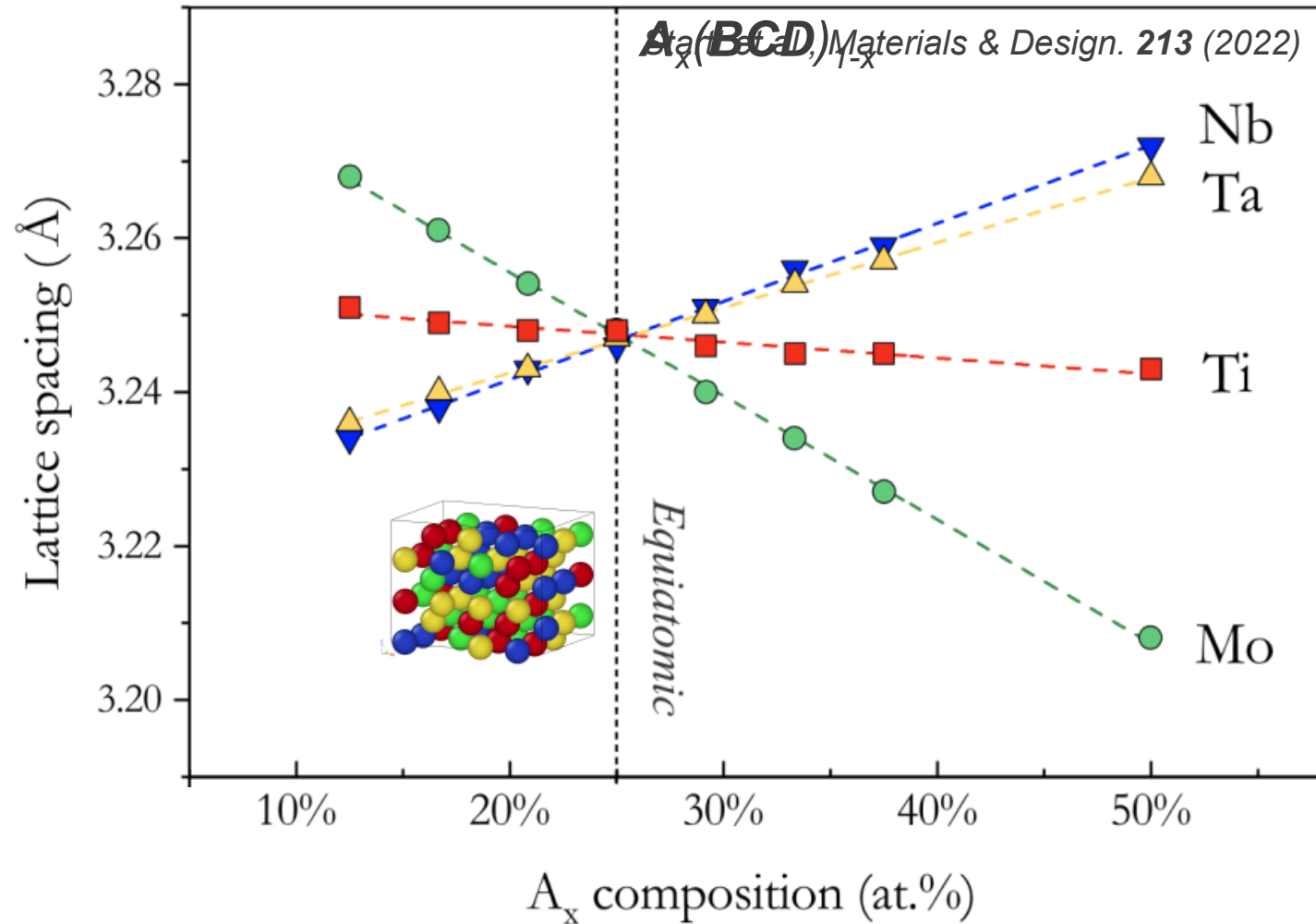
AKA: high-entropy alloy (HEA), multi-principal element alloy (MPEA)



Coury et al., Acta Mat. 175 (2019)

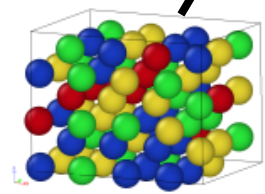
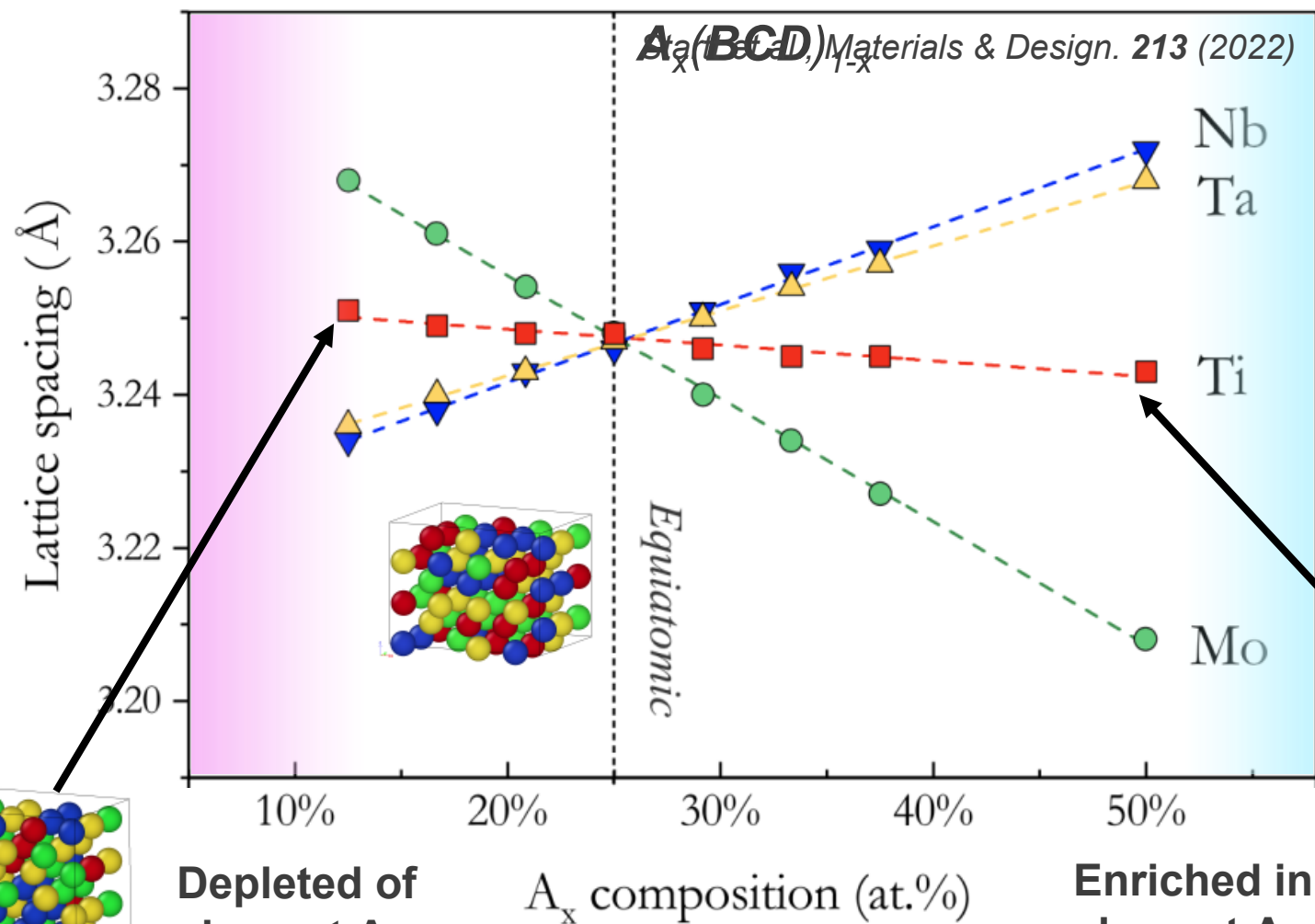
Exploring MoNbTaTi composition space - DFT

Vary element **A**, hold elements **B, C, and D**
constant



Exploring MoNbTaTi composition space - DFT

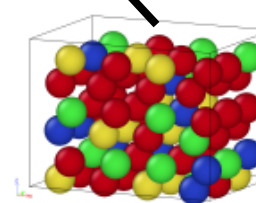
Vary element **A**, hold elements **B, C, and D**
constant



Depleted of
element A

A_x composition (at.%)

Enriched in
element A

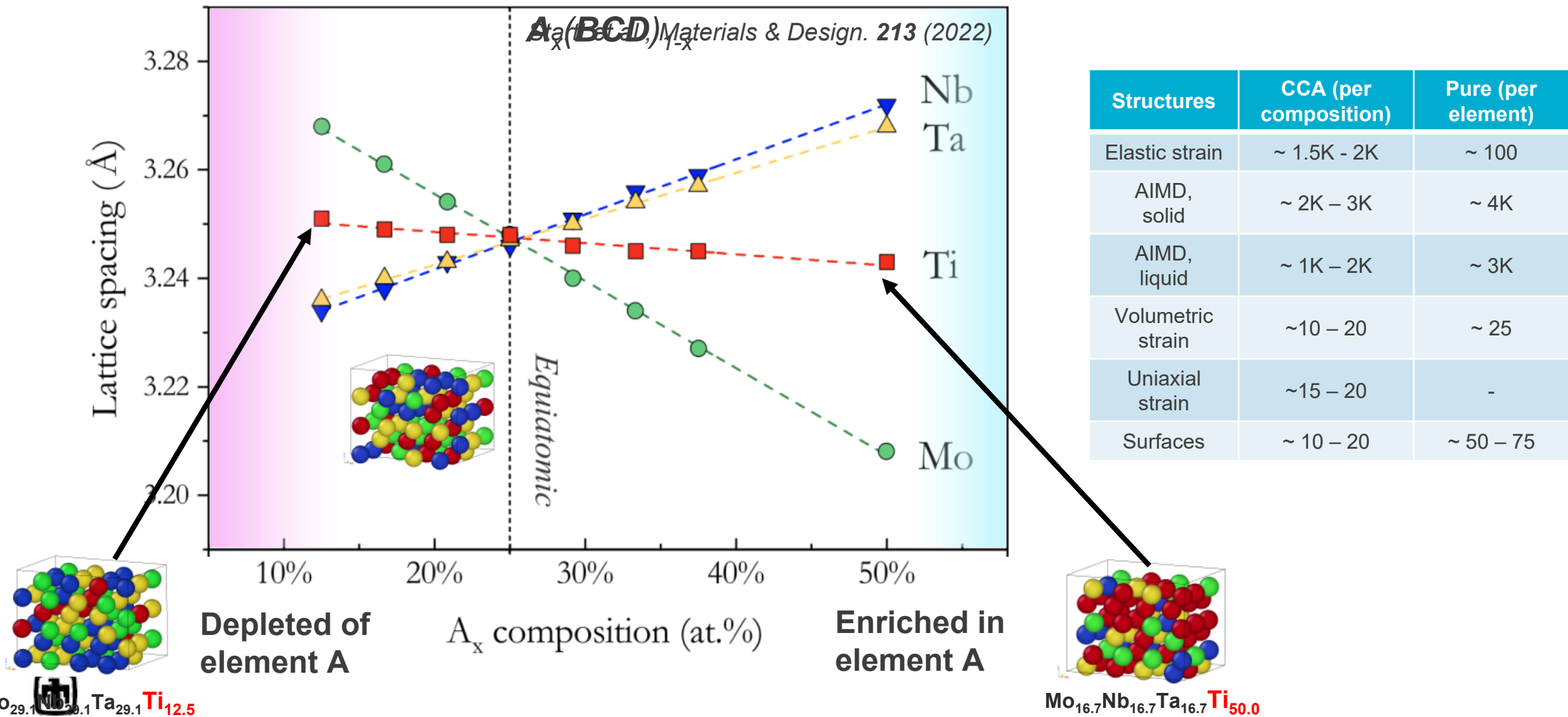


Mo_{16.7}Nb_{16.7}Ta_{16.7}Ti_{50.0}

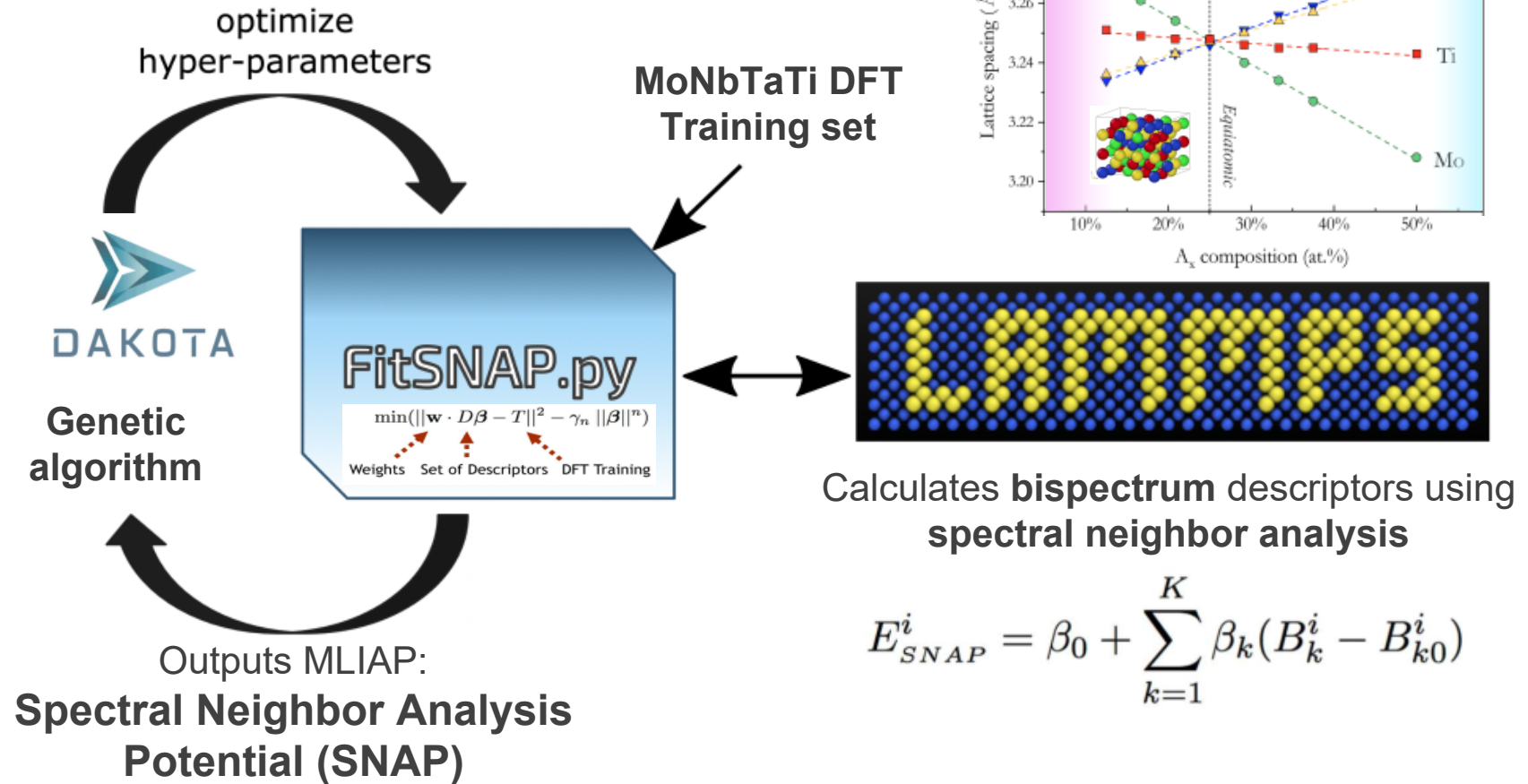
Mo_{29.1}Nb_{29.1}Ta_{29.1}Ti_{12.5}

8 Exploring MoNbTaTi composition space - DFT

Vary element *A*, hold elements *B*, *C*, and *D* constant



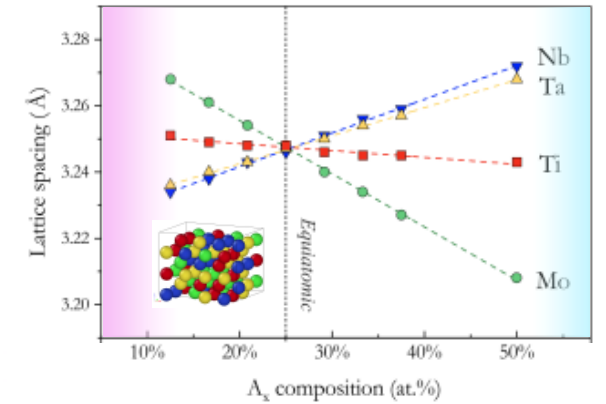
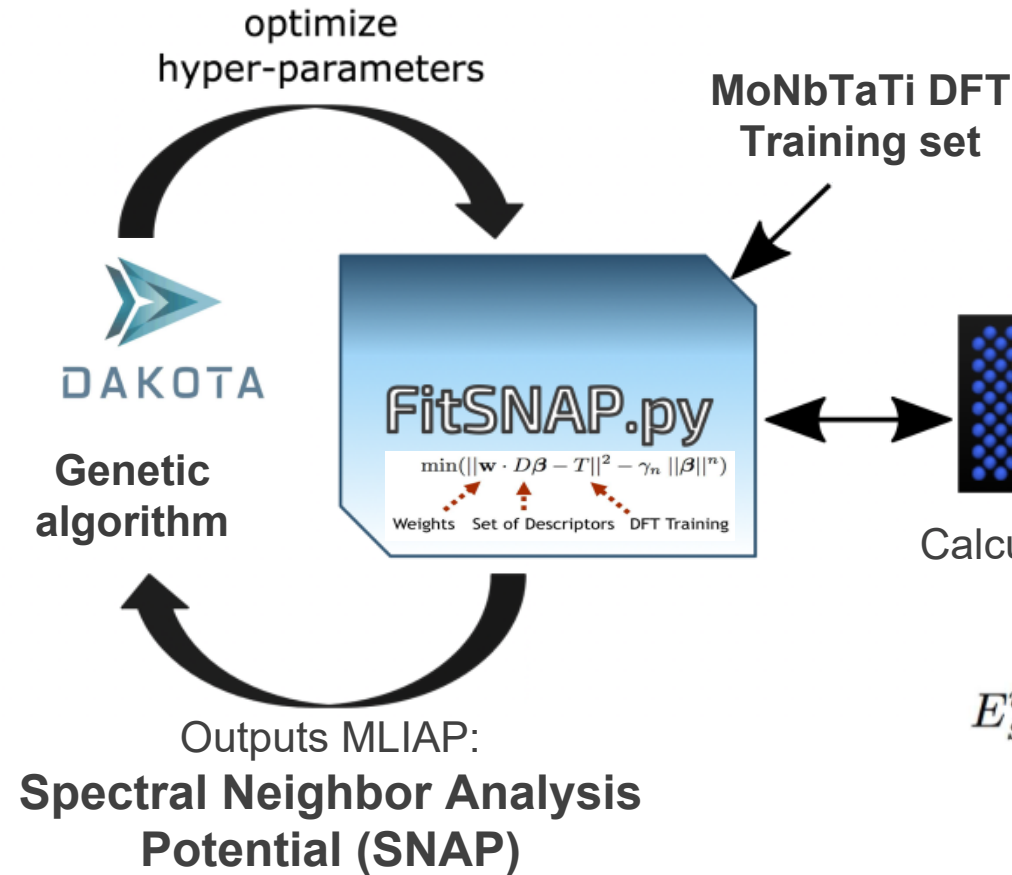
9 MLIAP training scheme



10 MLIAP training scheme

Goal 1: Understand ‘yield’ of rich composition-varied training set (no additional structures)

→ MLIAP are trained to match **trends in elastic properties** with changes in **composition**



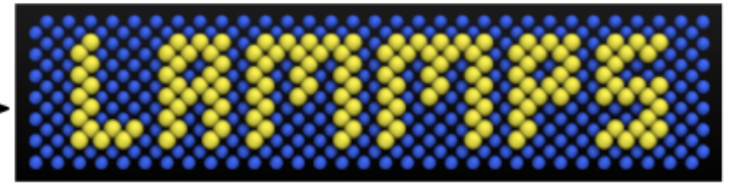
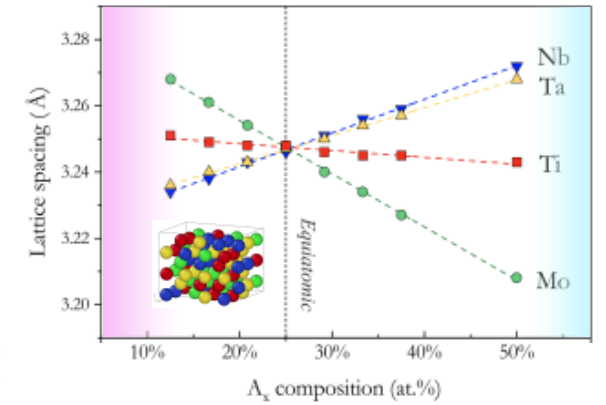
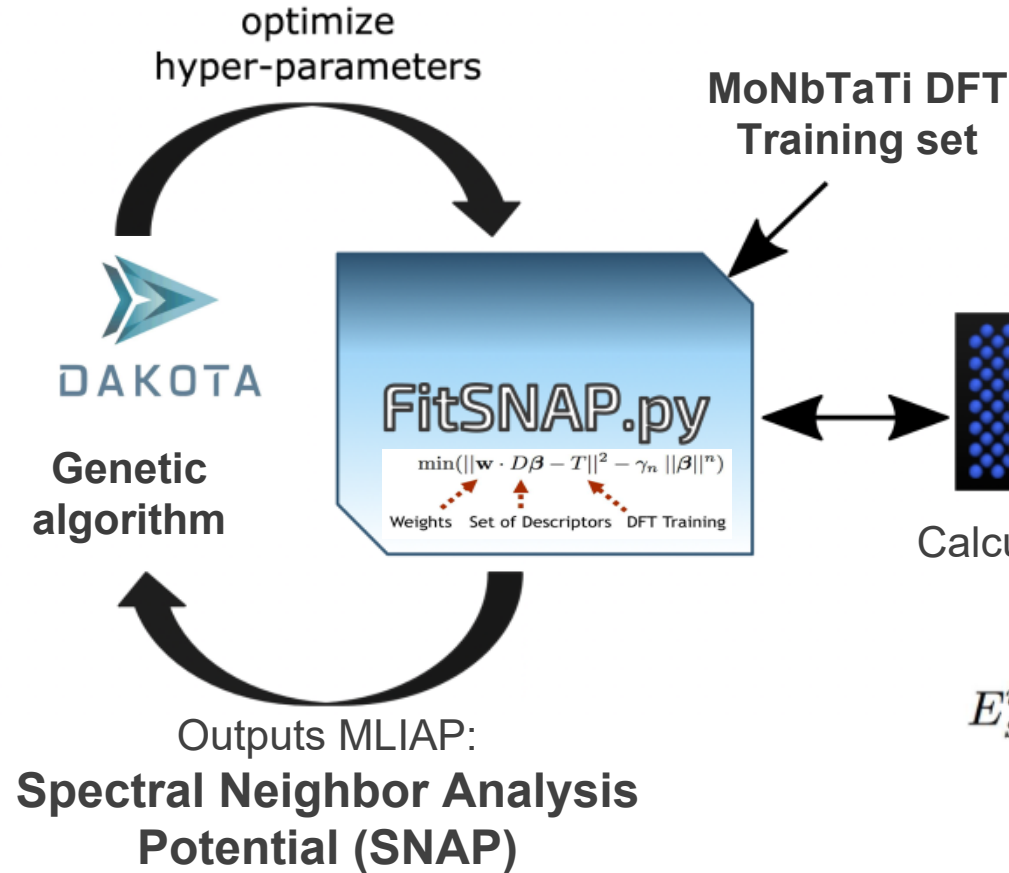
Calculates **bispectrum** descriptors using **spectral neighbor analysis**

$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_{k0}^i)$$

11 MLIAP training scheme

Goal 1: Understand ‘yield’ of rich composition-varied training set (no additional structures)

→ MLIAP are trained to match **trends in elastic properties** with changes in composition



Calculates **bispectrum** descriptors using **spectral neighbor analysis**

$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_{k0}^i)$$

$$E = \frac{9GB}{3B+G}$$



$$B_0 = \frac{C_{1111} + 2C_{1122}}{3}$$

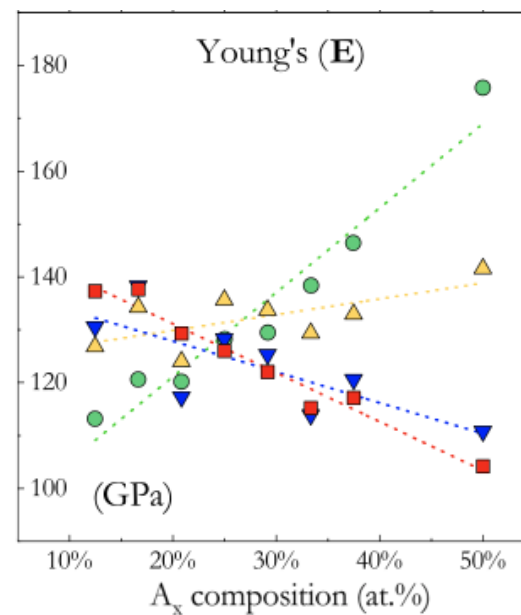
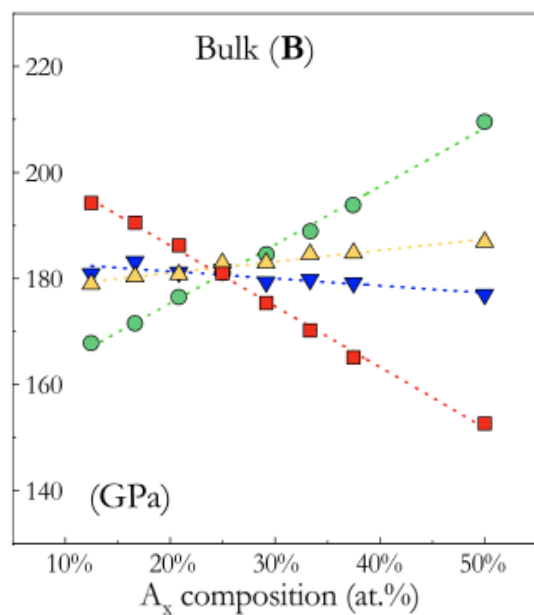
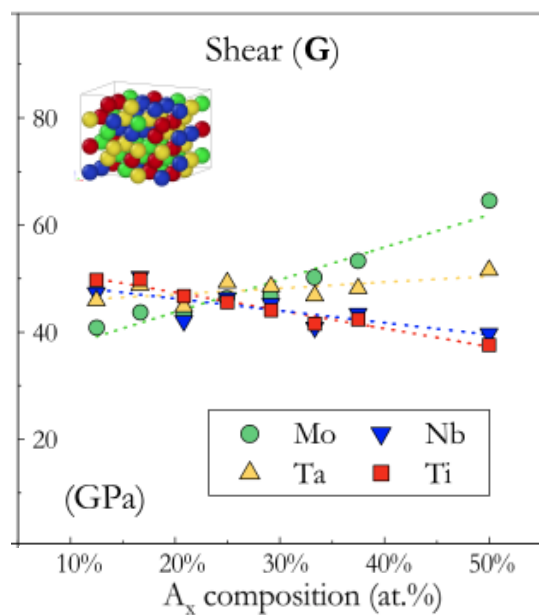
$$G = \frac{1}{2} \left[\frac{C_{1111} - C_{1122}}{5} + \frac{3C_{1212}}{5} + \frac{5C_{1212} - C_{1111} - C_{1122}}{4C_{1212} + 3(C_{1111} - C_{1122})} \right]$$



Example: Young's modulus $E \rightarrow$ Genetic algorithm favors MLIAP with low errors on cubic **C11**, **C12**, **C44** 11

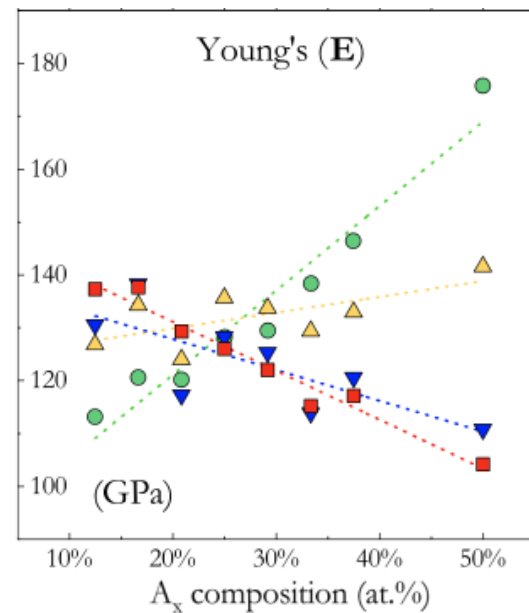
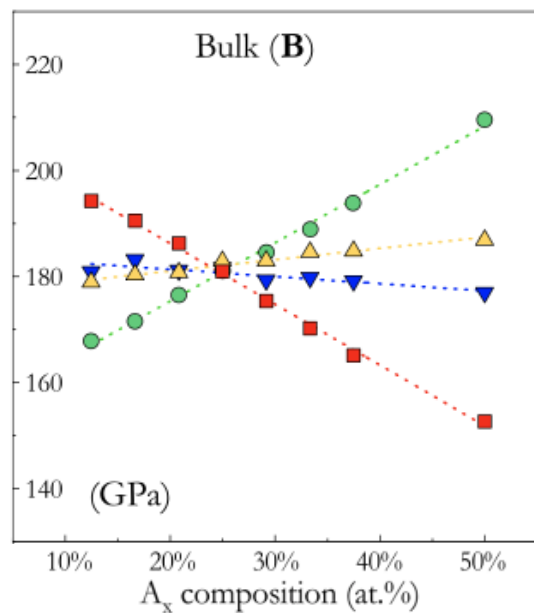
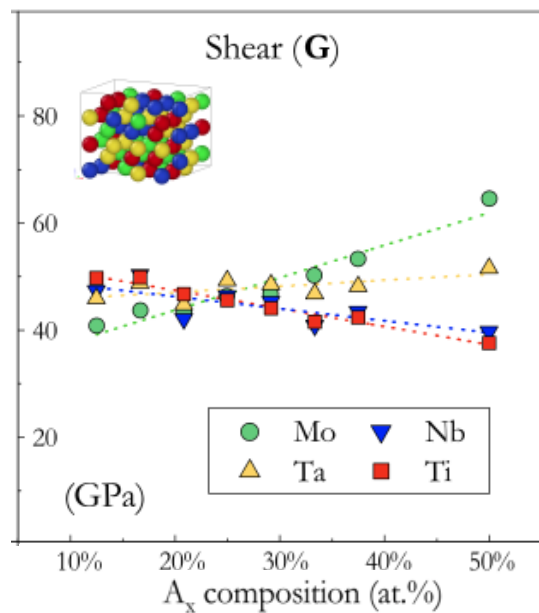
Fits to higher-order elastic properties

DFT



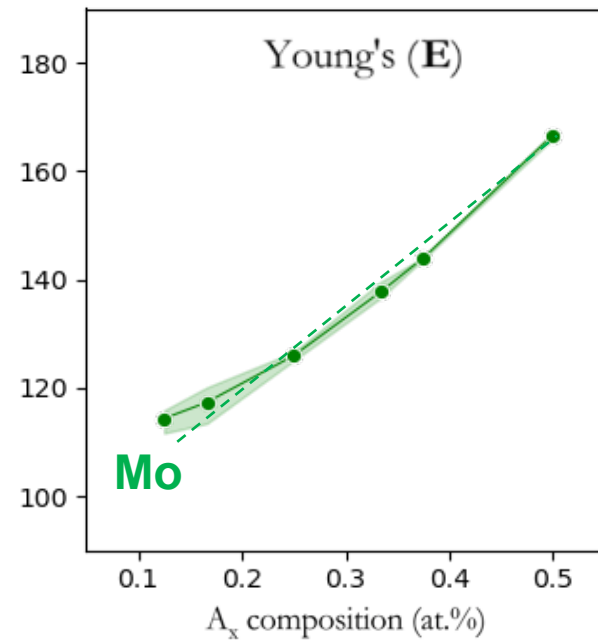
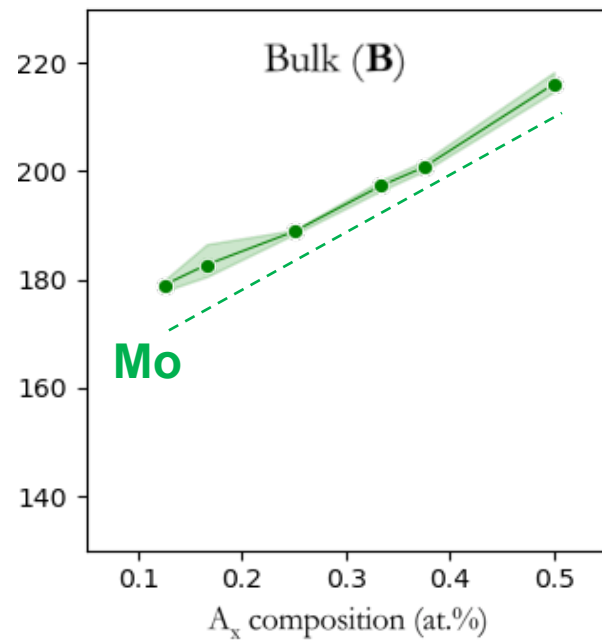
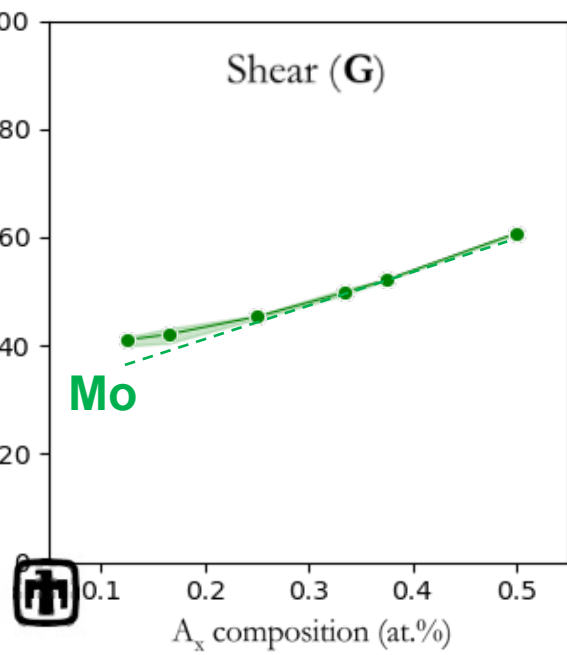
Fits to higher-order elastic properties

DFT



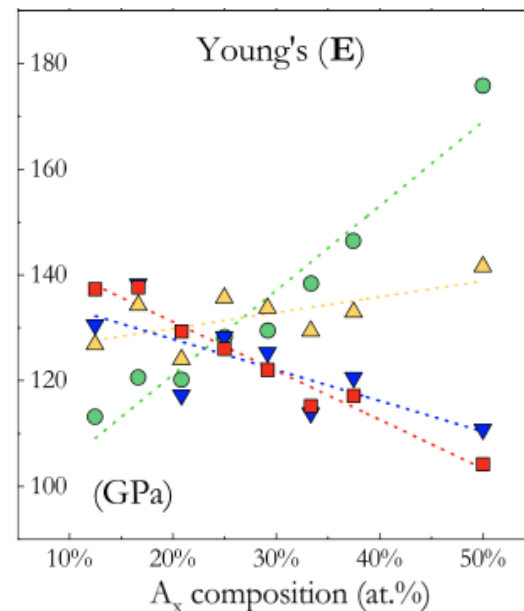
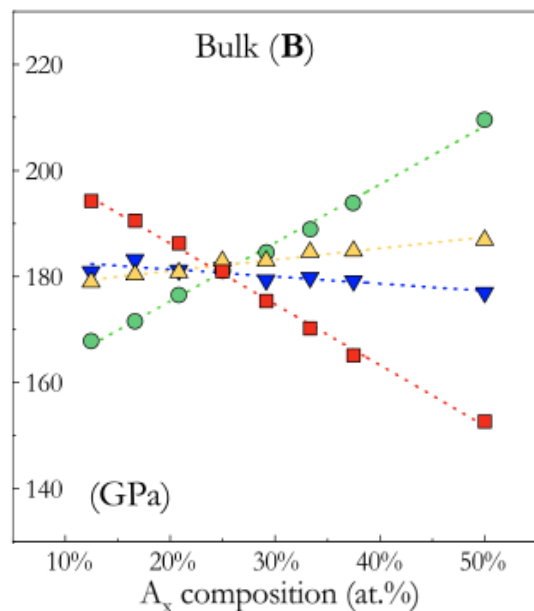
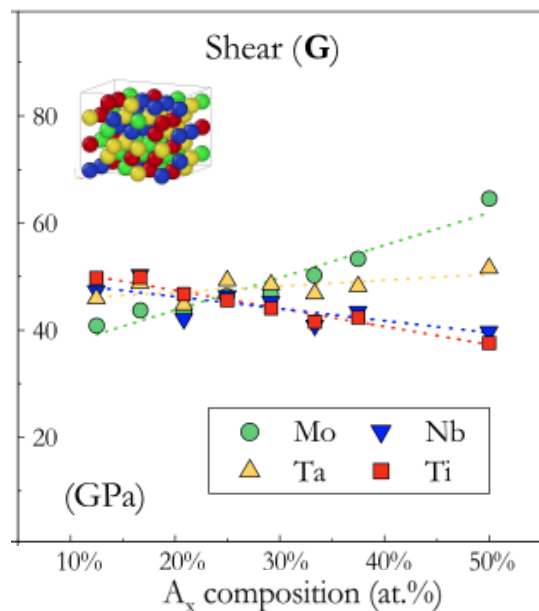
- Can reliably get stable MLIAPs that match DFT trends well across composition space for single elements
- Tested successfully at high T, P, and strain rates

SNAP

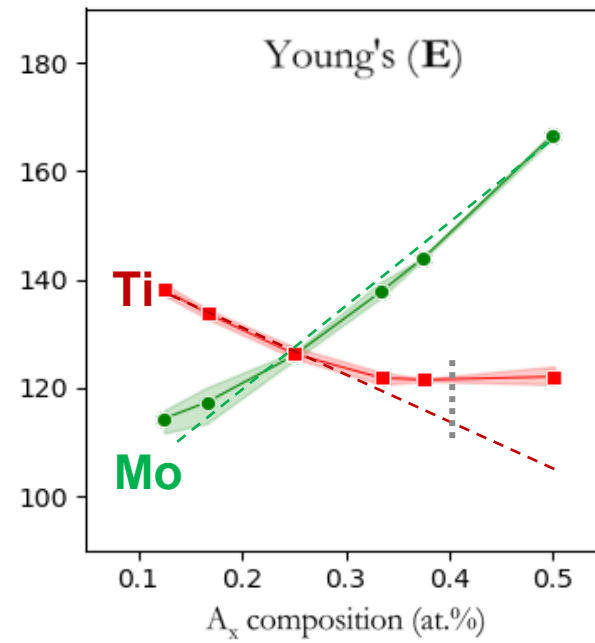
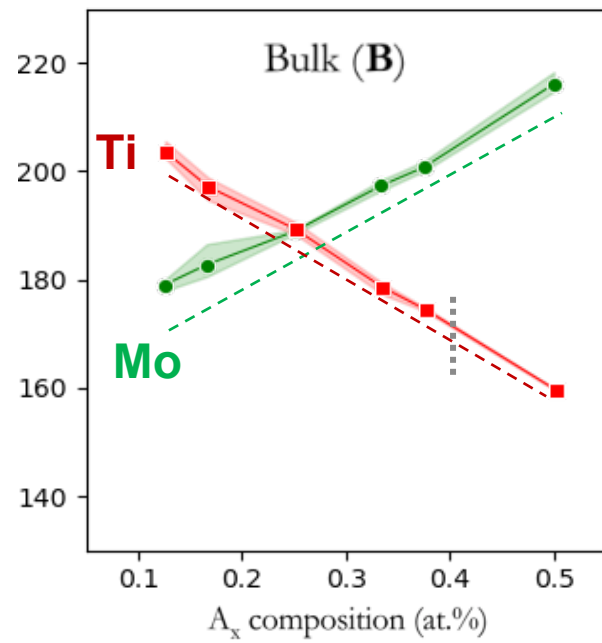
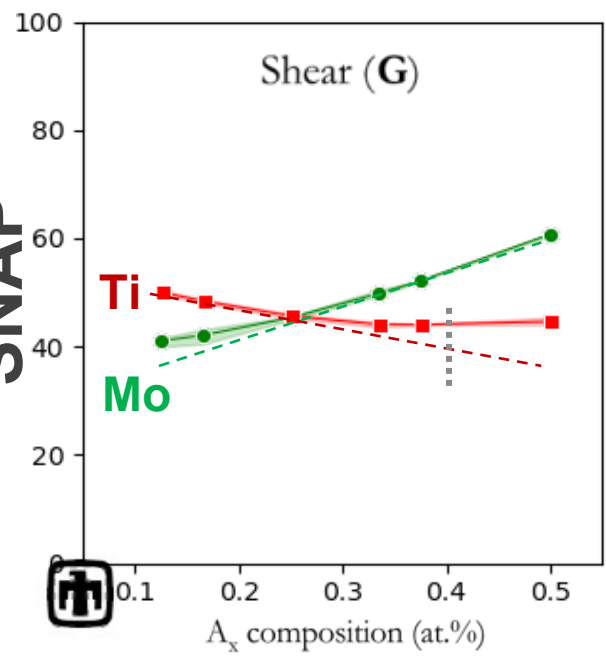


Fits to higher-order elastic properties

DFT



SNAP



- Can reliably get stable MLIAPs that match DFT trends well across composition space for single elements
- Tested successfully at high T, P, and strain rates
- Now beginning to get stable MLIAP that can match **two elements** → can **extrapolate** to new compositions! (here: most Mo at-%, up to Ti ~35 %)

Extrapolation from training set

Untrained composition (vary 2 elements)



Property	SNAP 2-element MLIAP	New DFT (not in training)	SNAP - DFT
C11 (GPa)	239.6	237.6	0.8 %
C12 (GPa)	143.6	129.7	10.7 %
C44 (GPa)	39.5	37.8	4.5 %
B (GPa)	175.6	165.7	6.0 %
G (GPa)	42.7	43.6	2.1 %
E (GPa)	118.5	120.1	1.3 %

Startt et al., Materials & Design, 213 (2022)



Extrapolation from training set

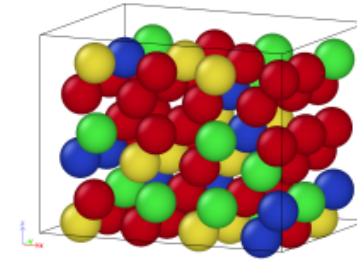
Untrained composition (vary 2 elements)



Property	SNAP 2-element MLIAP	New DFT (not in training)	SNAP - DFT
C11 (GPa)	239.6	237.6	0.8 %
C12 (GPa)	143.6	129.7	10.7 %
C44 (GPa)	39.5	37.8	4.5 %
B (GPa)	175.6	165.7	6.0 %
G (GPa)	42.7	43.6	2.1 %
E (GPa)	118.5	120.1	1.3 %

Startt et al., Materials & Design, 213 (2022)

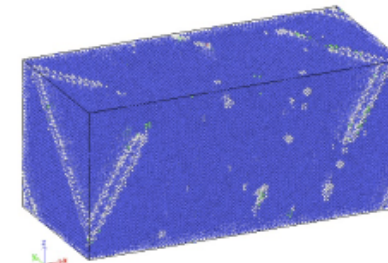
Increase system size



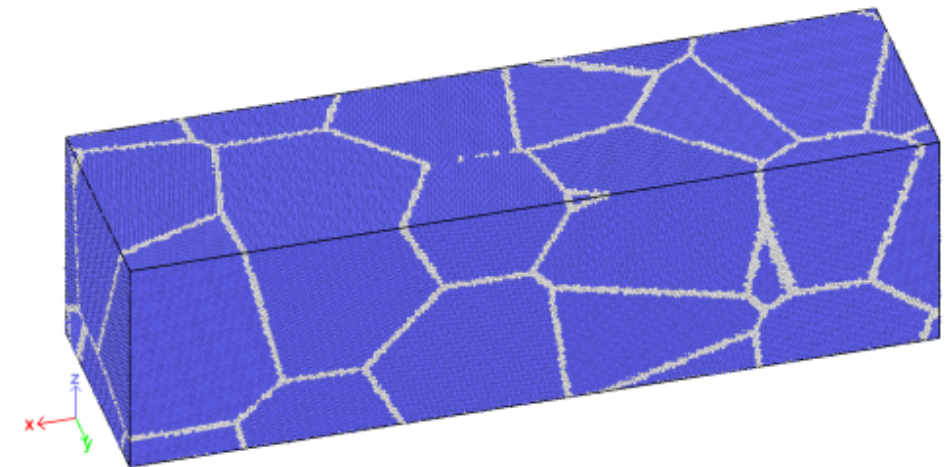
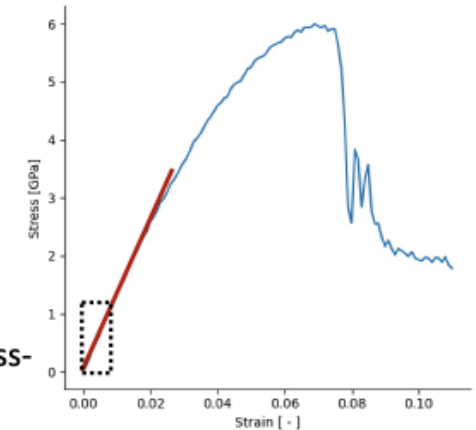
• SNAP | DFT

72-atom cell

$\epsilon = 6.25\%$



Young's Modulus from stress-strain curve: ~120 GPa



Extrapolation from training set

Untrained composition (vary 2 elements)

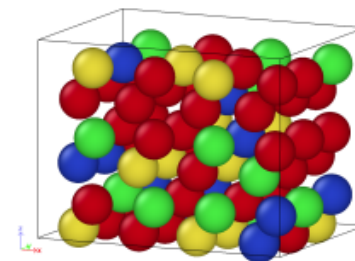


Property	SNAP 2-element MLIAP	New DFT (not in training)	SNAP - DFT
C11 (GPa)	239.6	237.6	0.8 %
C12 (GPa)	143.6	129.7	10.7 %
C44 (GPa)	39.5	37.8	4.5 %
B (GPa)	175.6	165.7	6.0 %
G (GPa)	42.7	43.6	2.1 %
E (GPa)	118.5	120.1	1.3 %

Startt et al., Materials & Design, 213 (2022)

FitSNAP →

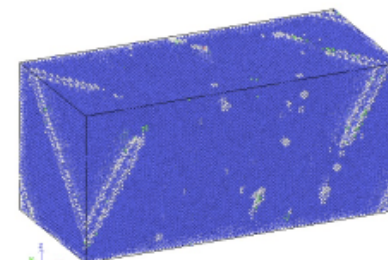
Active learning (AL) and
uncertainty quantification (UQ)
modules in development!



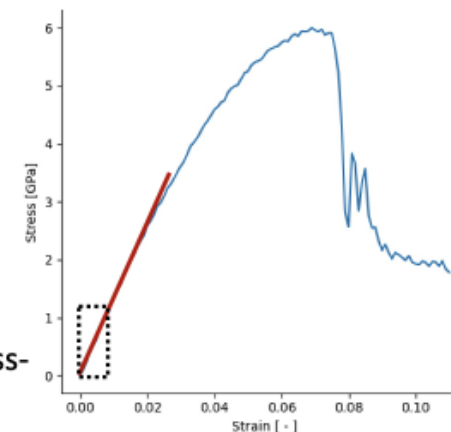
• SNAP | DFT

72-atom cell

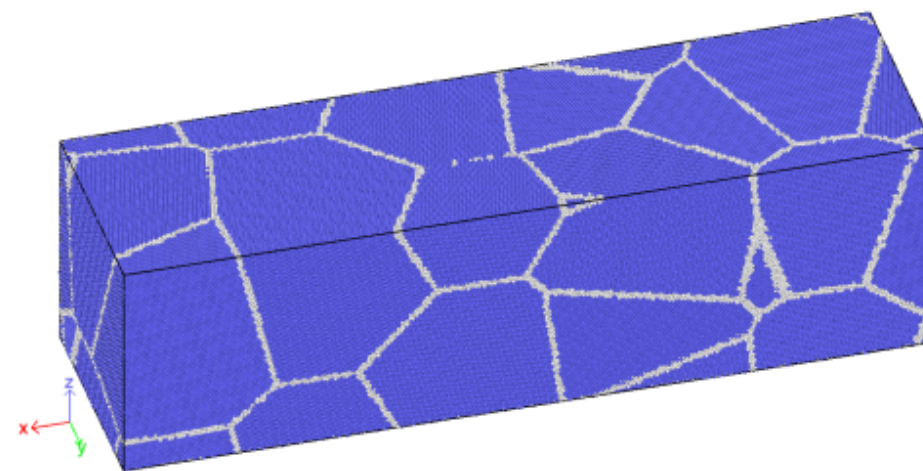
$\epsilon = 6.25\%$



Young's Modulus from stress-strain curve: ~120 GPa



Increase system size



Sample new MD environments, add DFT data

Building the training set beyond elasticity



- Multiple compositions – explosion in number of training structures to include in DFT (expensive!)
 - Current MoNbTaTi comprehensive DFT set: **no point defect structures** or **derivative (binary, ternary) alloys**
- Use AL/UQ studies of **point defect behavior** and **alloy chemistry** to expand training set *where necessary*



Building the training set beyond elasticity



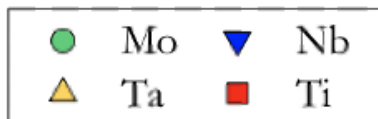
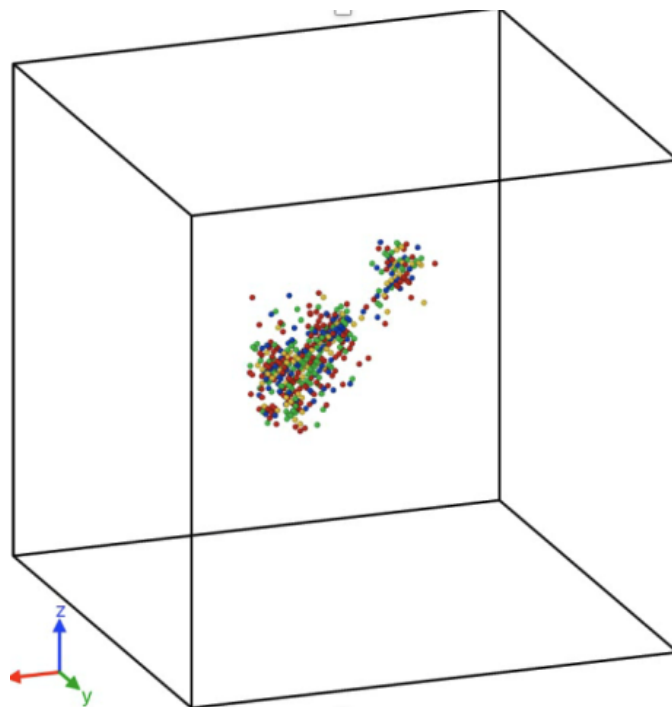
- Multiple compositions – explosion in number of training structures to include in DFT (expensive!)
 - Current MoNbTaTi comprehensive DFT set: **no point defect structures** or **derivative (binary, ternary) alloys**
- Use AL/UQ studies of **point defect behavior** and **alloy chemistry** to expand training set *where necessary*

Point defects: Radiation damage

Primary knock-on
atom (PKA)
simulations →

Expose SNAP to per-
atom energy levels
far exceeding range
seen in training set

Training: order of
~10 eV
PKA:
5 keV, 10 keV, 20
keV



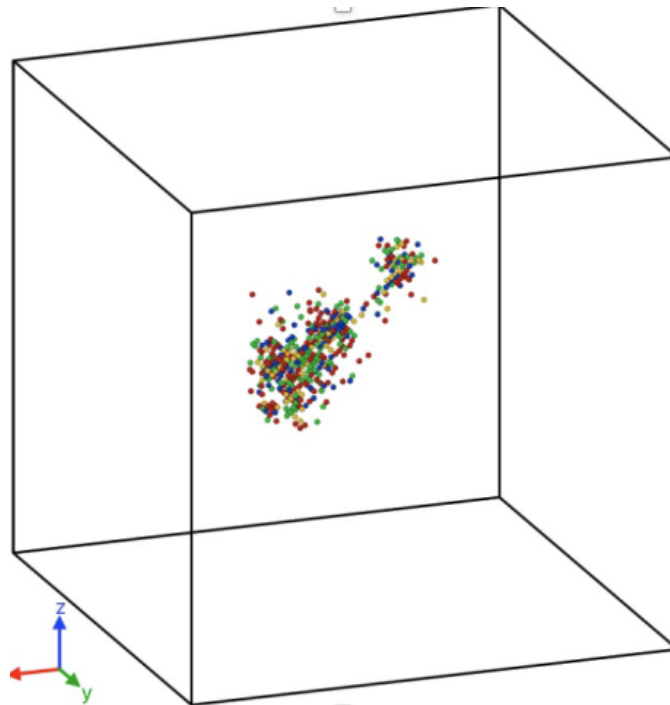
Building the training set beyond elasticity



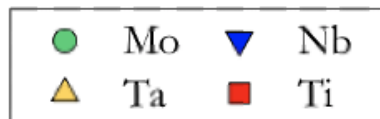
- Multiple compositions – explosion in number of training structures to include in DFT (expensive!)
 - Current MoNbTaTi comprehensive DFT set: **no point defect structures** or **derivative (binary, ternary) alloys**
- Use AL/UQ studies of **point defect behavior** and **alloy chemistry** to expand training set *where necessary*

Point defects: Radiation damage

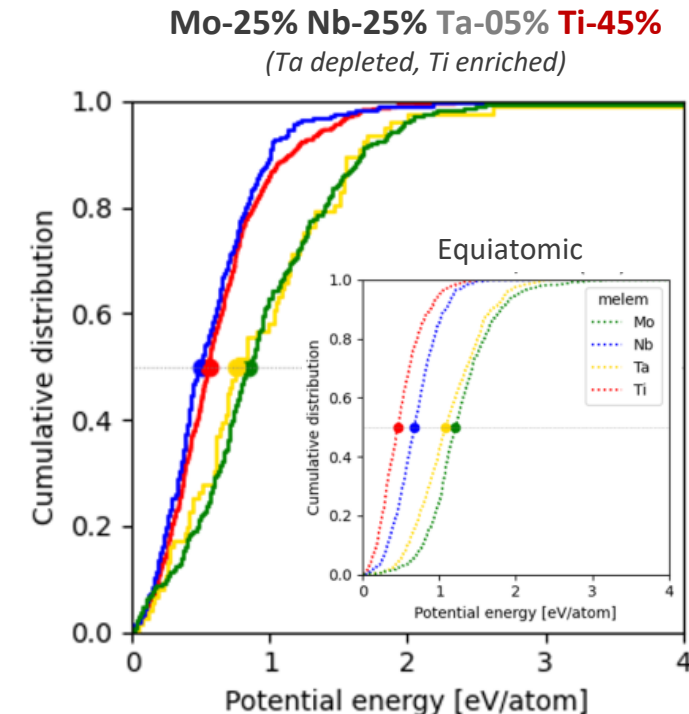
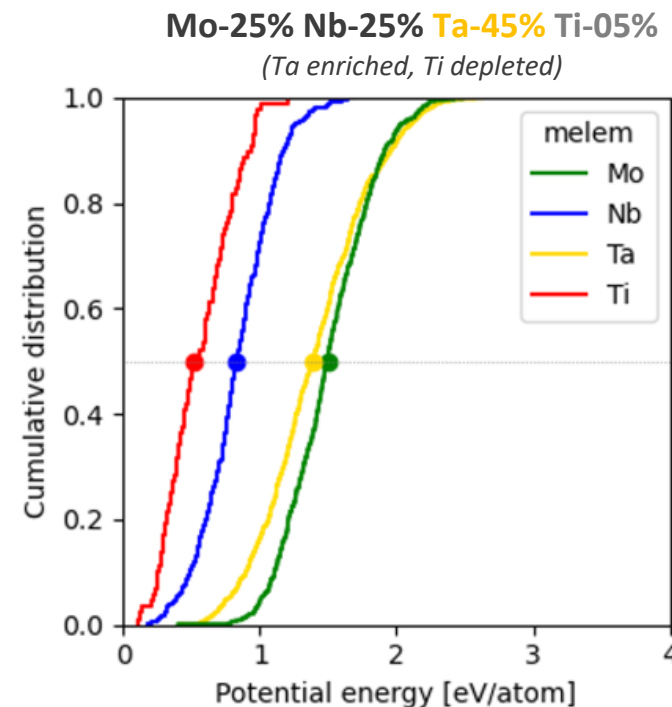
Primary knock-on atom (PKA) simulations →
Expose SNAP to per-atom energy levels far exceeding range seen in training set



Training: order of
~10 eV
PKA:
5 keV, 10 keV, 20 keV



Vacancy migration energy distributions (from NEB)

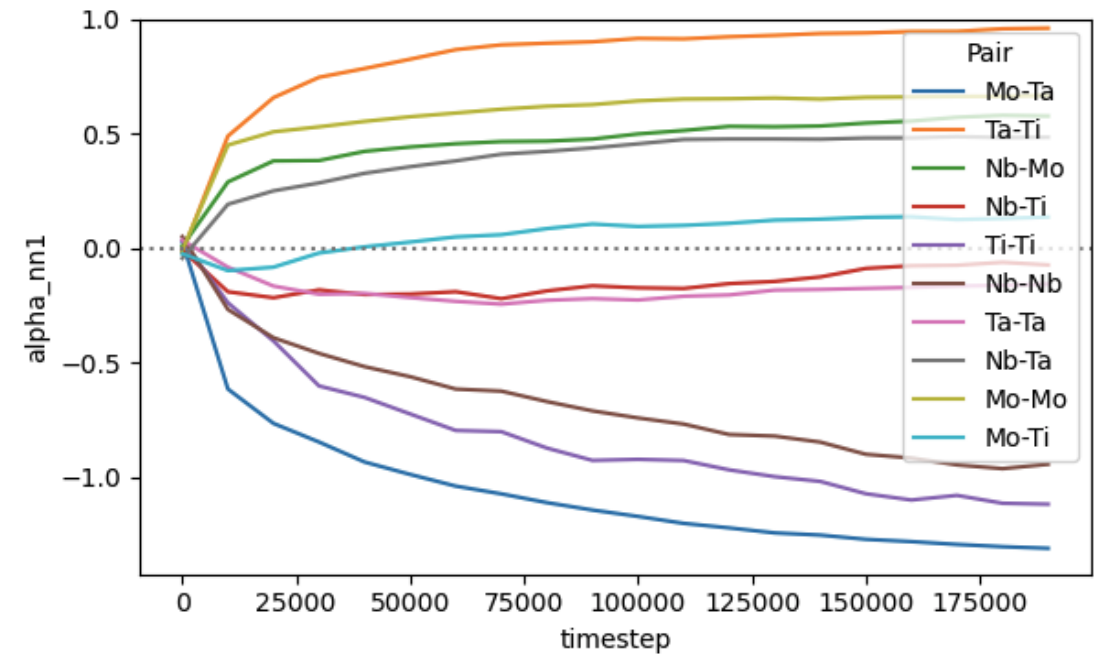
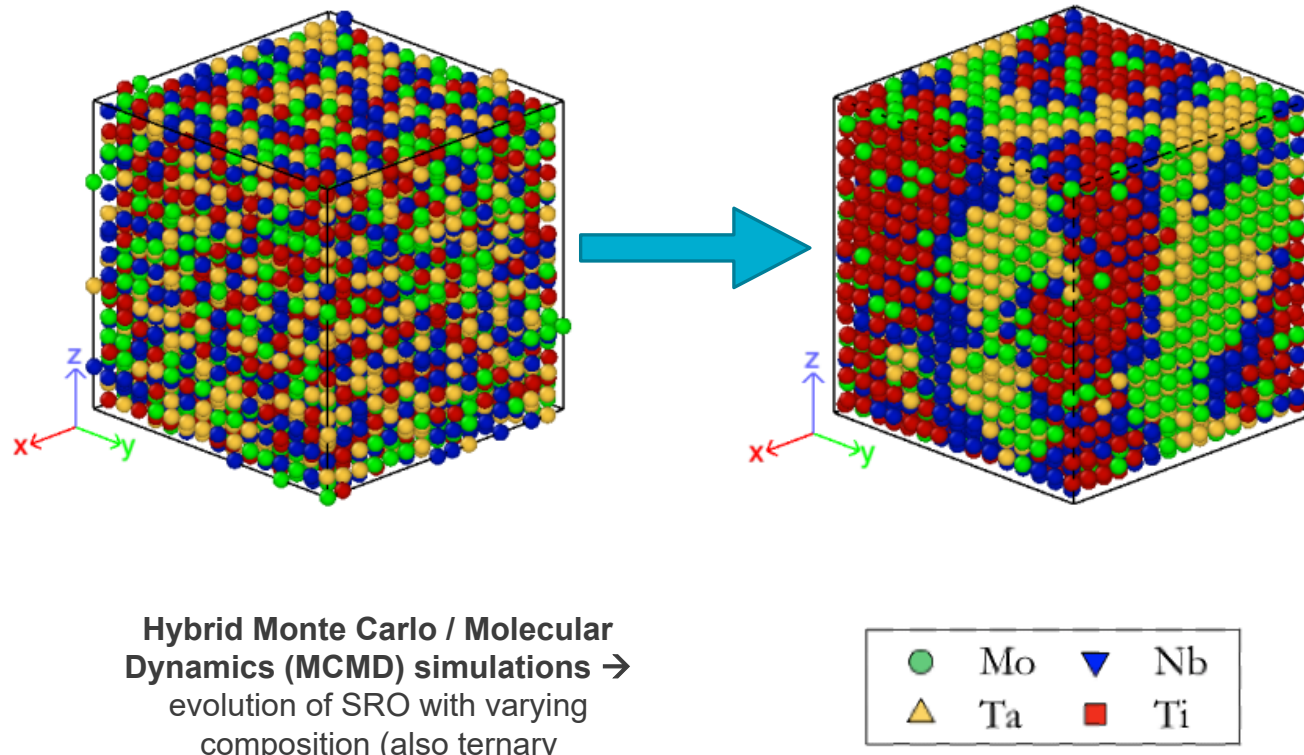


Building the training set beyond elasticity

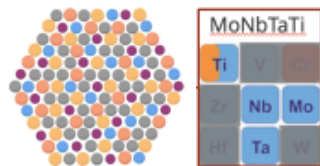


- Multiple compositions – explosion in number of training structures to include in DFT (expensive!)
 - Current MoNbTaTi comprehensive DFT set: **no point defect structures** or **derivative (binary, ternary) alloys**
- Use AL/UQ studies of **point defect behavior** and **alloy chemistry** to expand training set *where necessary*

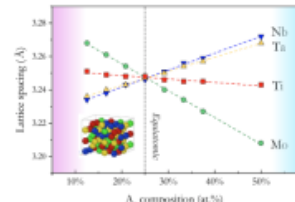
Alloy chemistry: short-range order studies



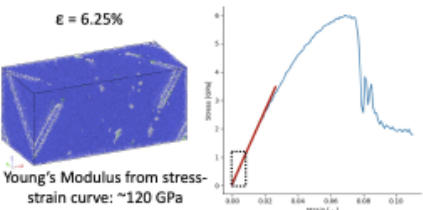
22 Summary



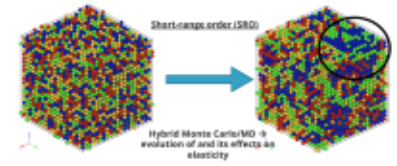
MoNbTaTi is a promising member of the refractory complex concentrated alloys



A comprehensive DFT study was undertaken by Startt et al. to understand how non-equiatomc composition affects material properties such as elasticity



Using FitSNAP and a genetic algorithm on the as-received DFT training set, we are successfully generating stable MLIAPs that can match elastic property trends across multiple compositions and extrapolate successfully to new compositions



Active learning and uncertainty quantification will be used to understand and expand scope of current MoNbTaTi training set



Contact:
megmcca@sandia.gov₂₂