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# Exploring the behavior of MoNbTaTi refractory CCAs across composition space using a machine-learned interatomic potential

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## Multiscale Materials Modeling (MMM10)

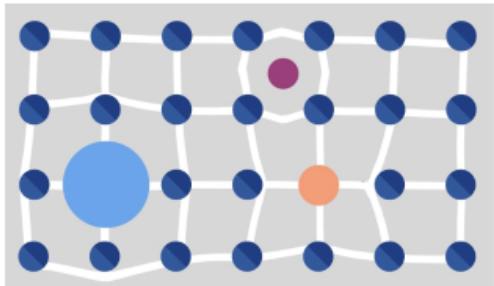
October 6<sup>th</sup>, 2022



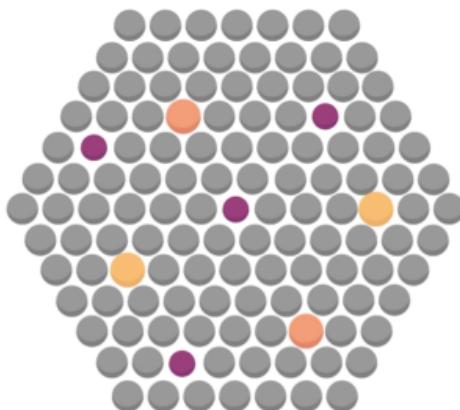
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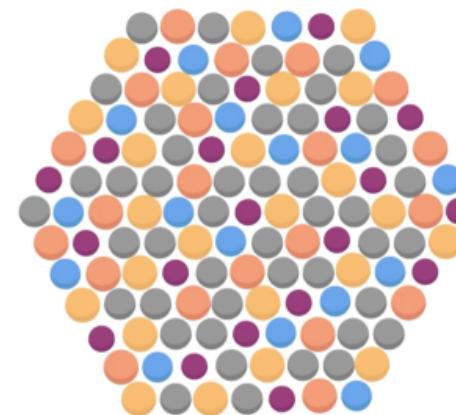
## Complex concentrated alloys



<https://doi.org/10.21203/rs.2.15081/v1>



Conventional alloy

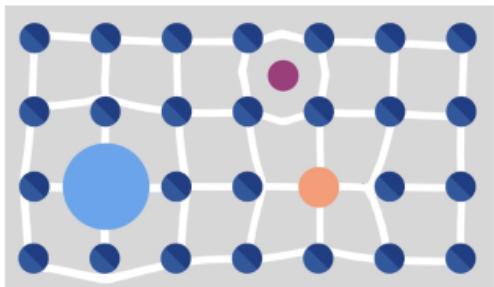


High-entropy alloy

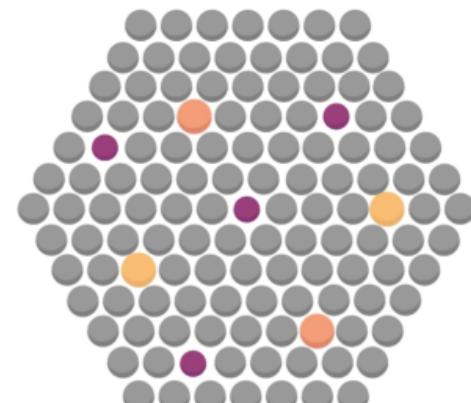
CCA: Alloy with high concentrations of 3 or more elements that can coexist without extreme phase separation

AKA: *high-entropy alloy (HEA)*, *multi-principal element alloy (MPEA)*

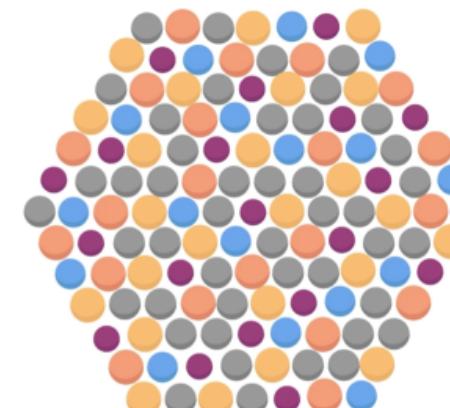
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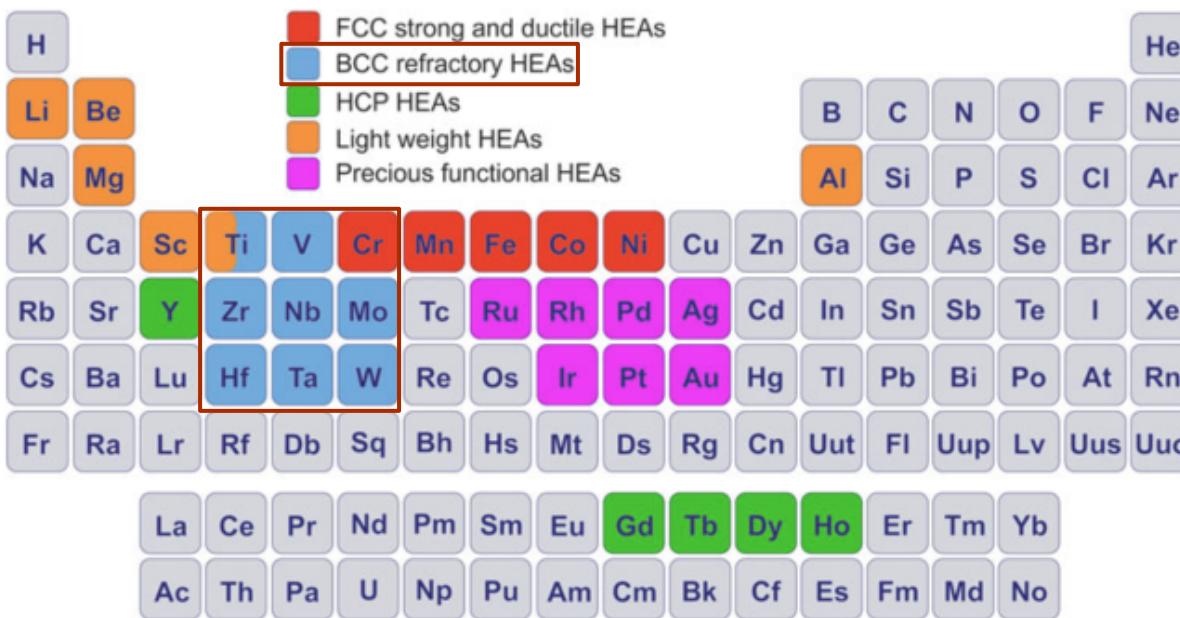
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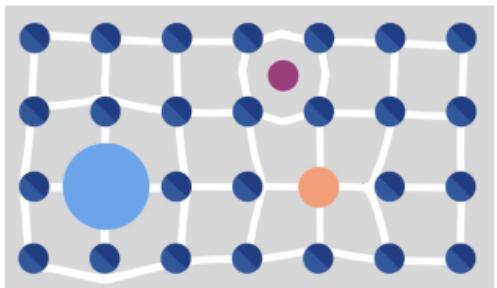
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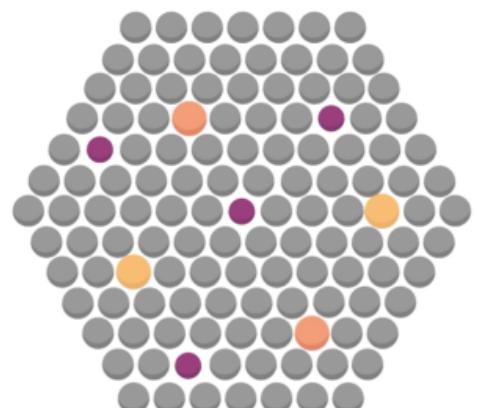
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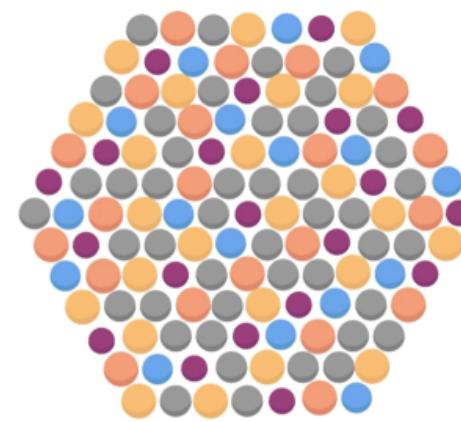
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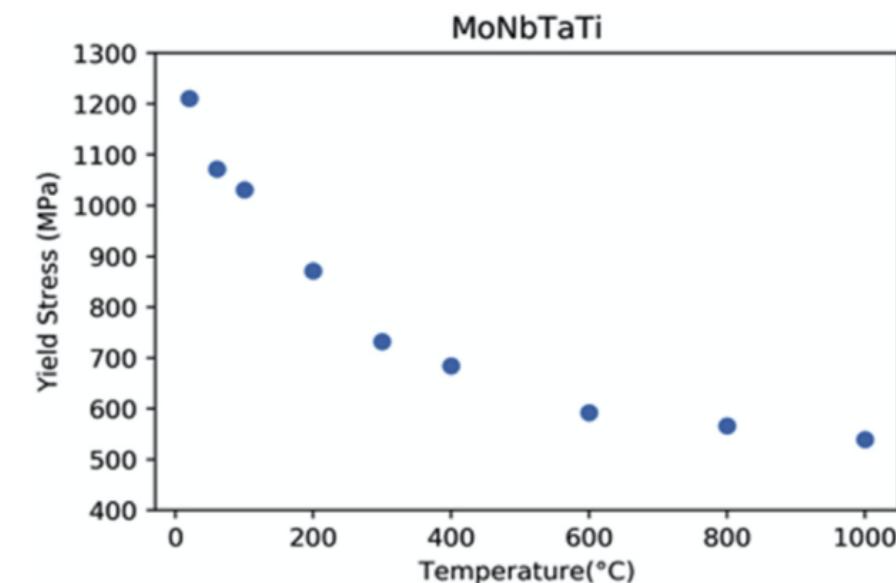
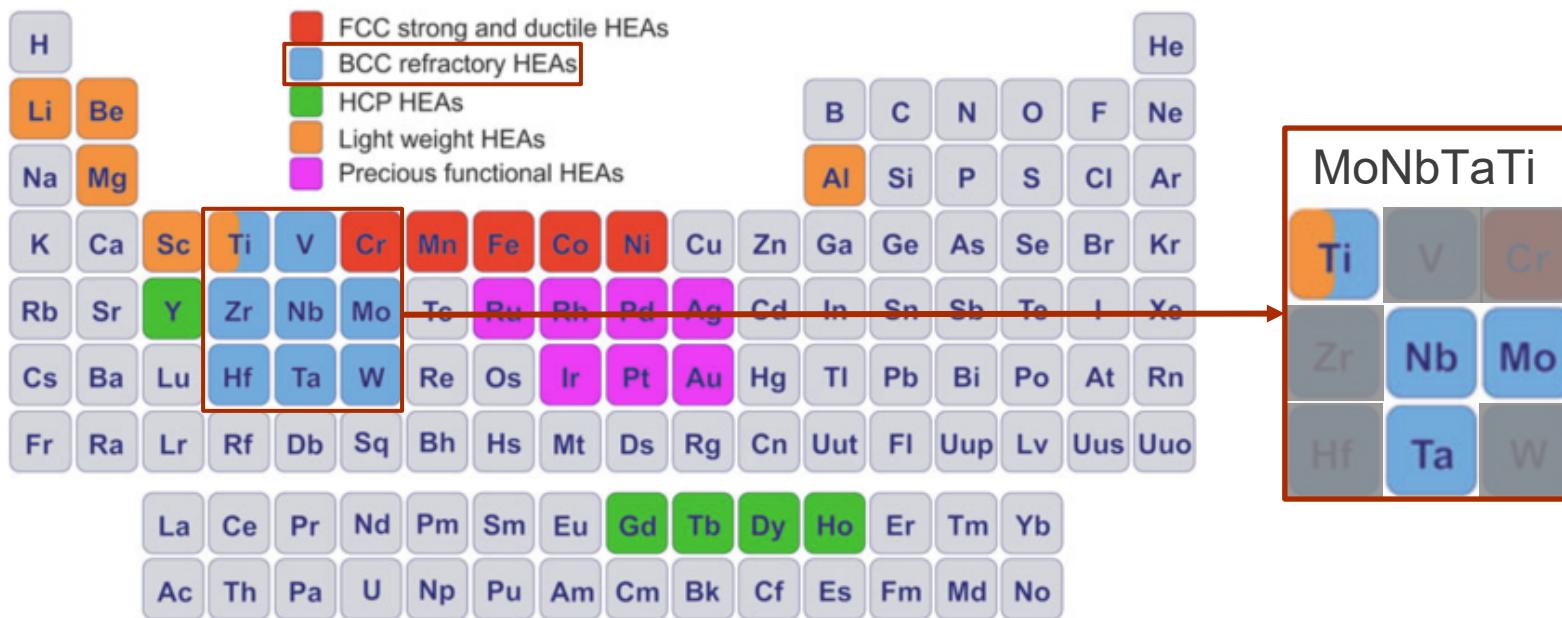
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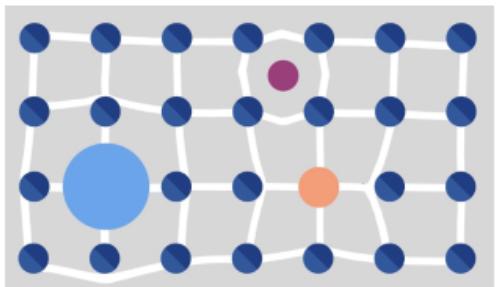
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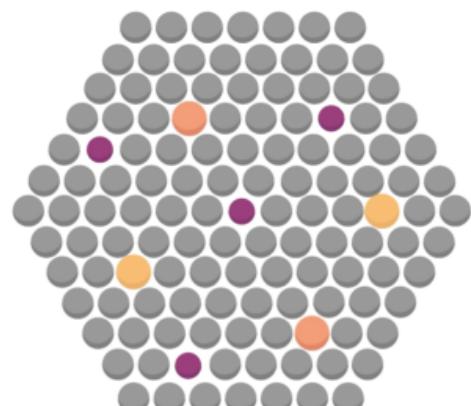


Coury et al., *Acta Mat.* 175 (2019)

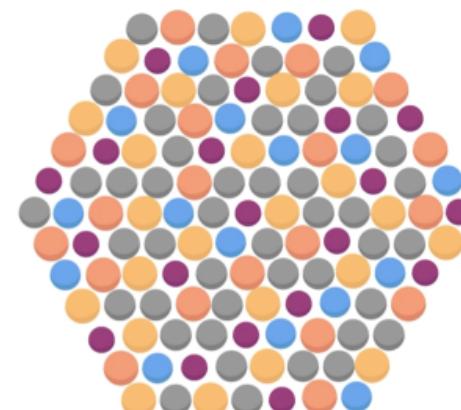
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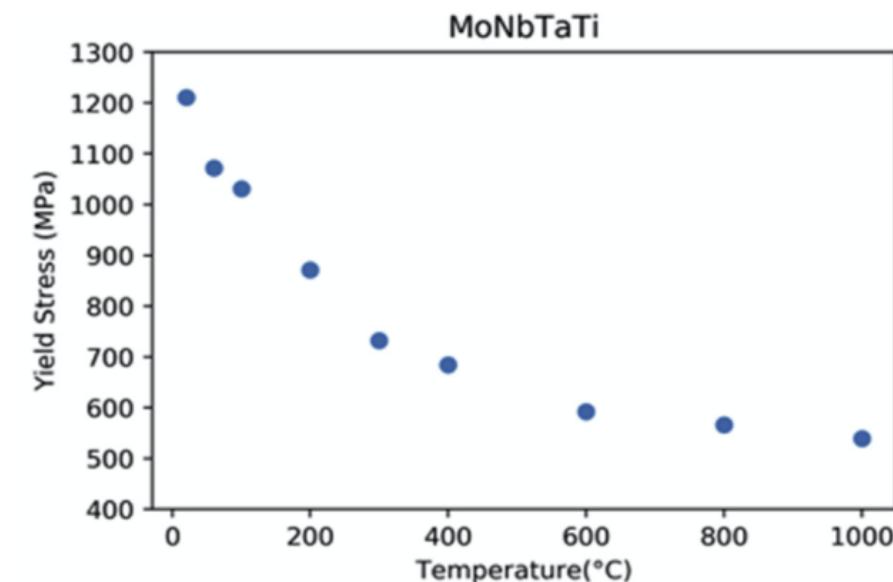
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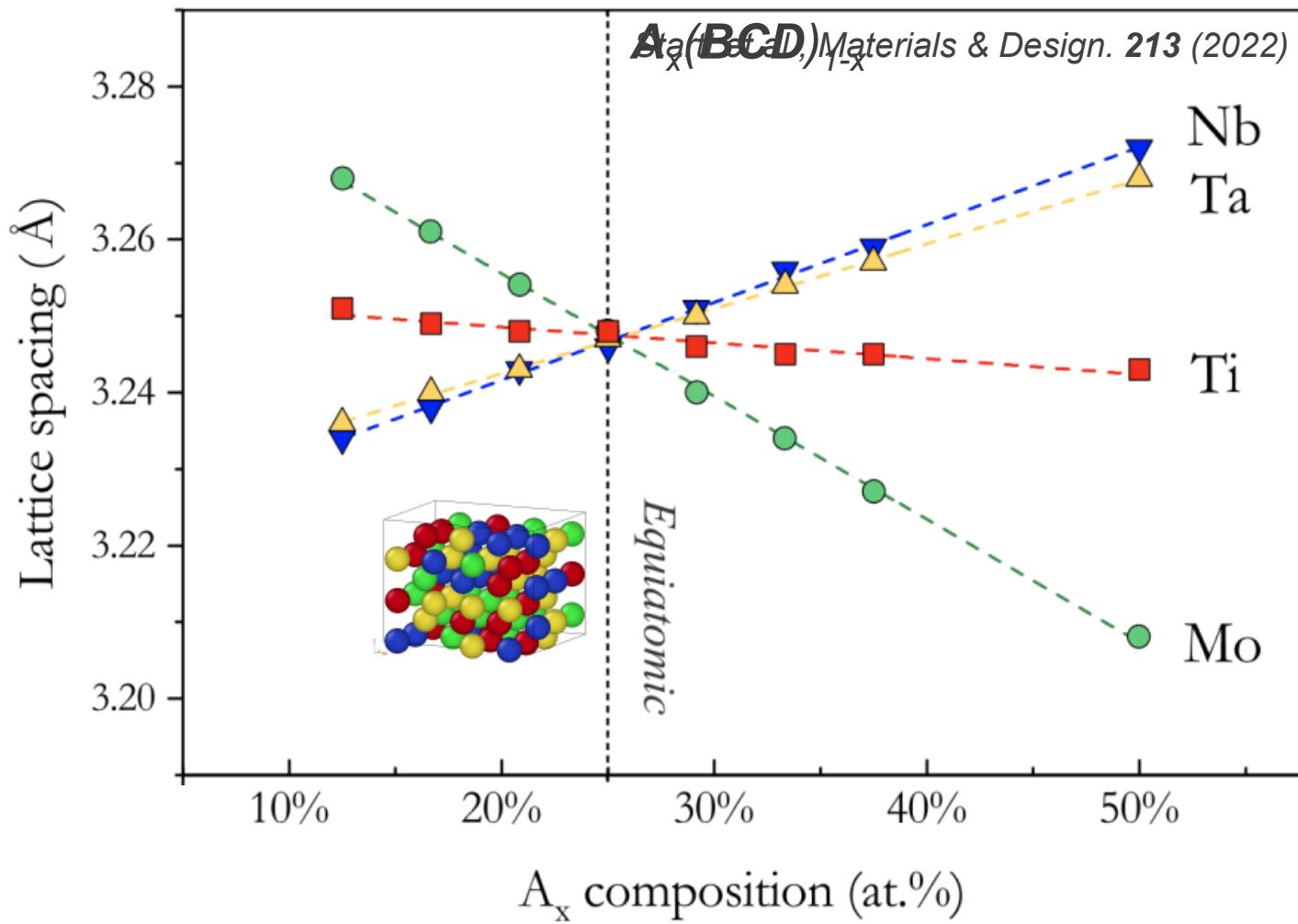
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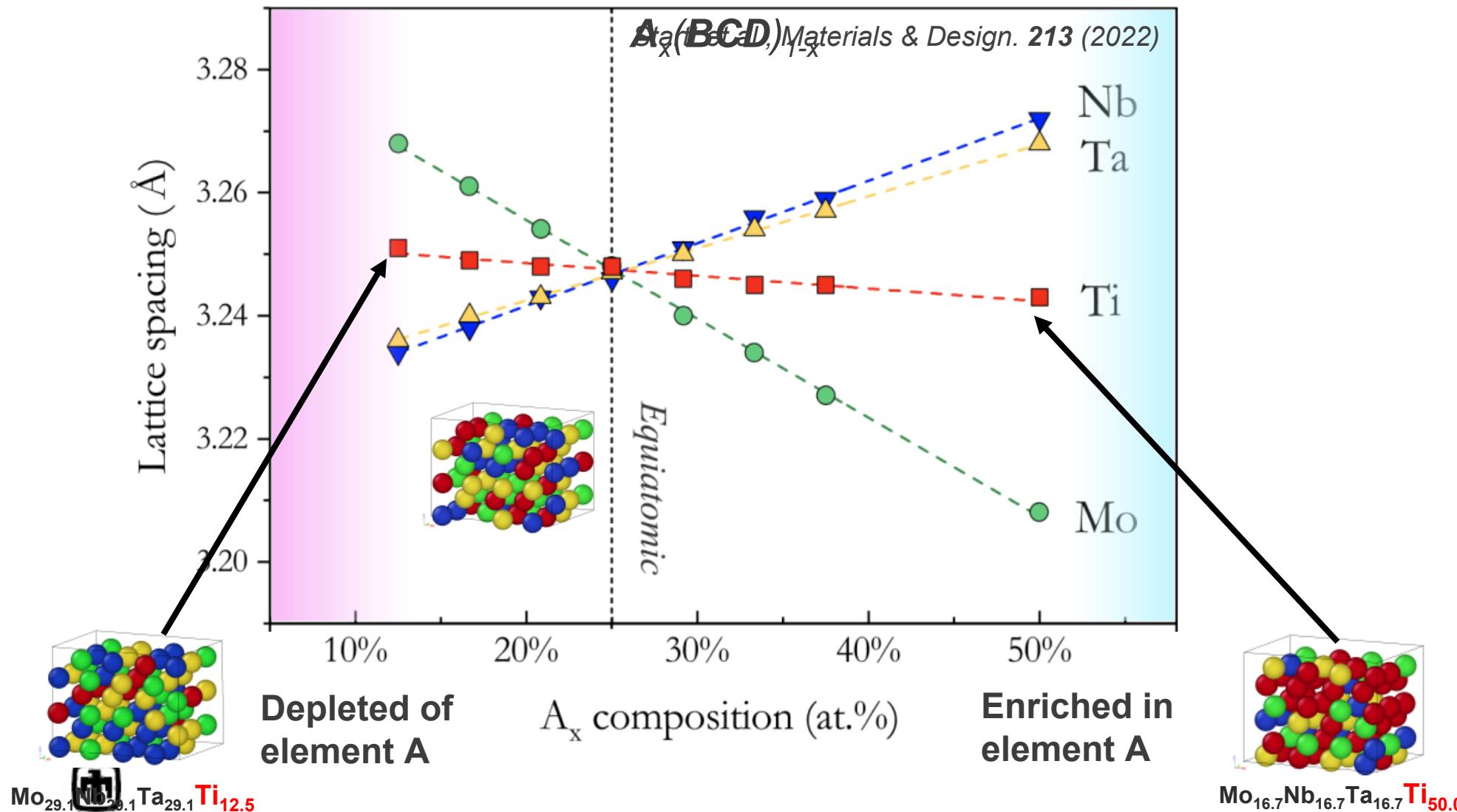
# Exploring MoNbTaTi composition space - DFT

*Vary element **A**, hold elements **B**, **C**, and **D** constant*



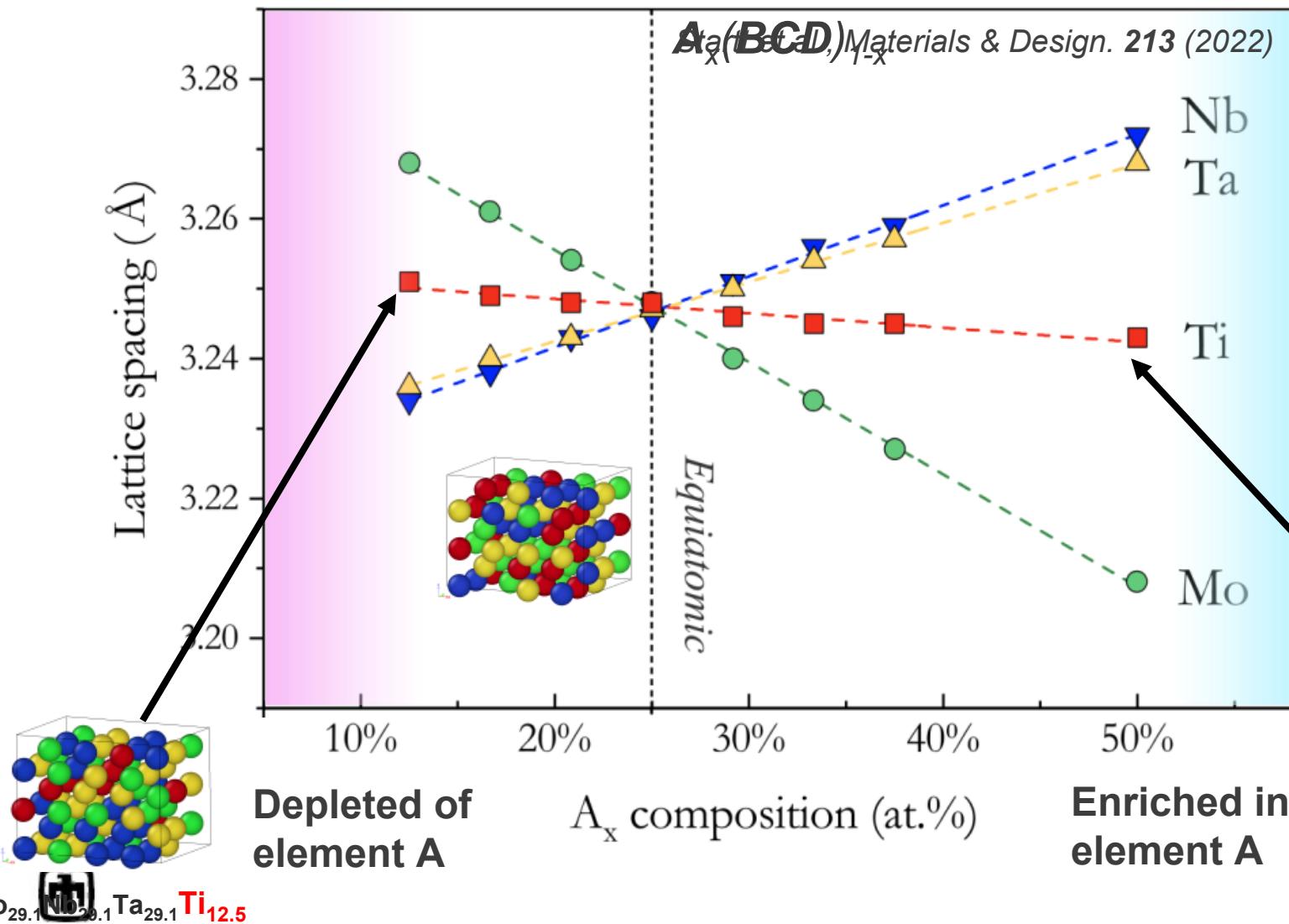
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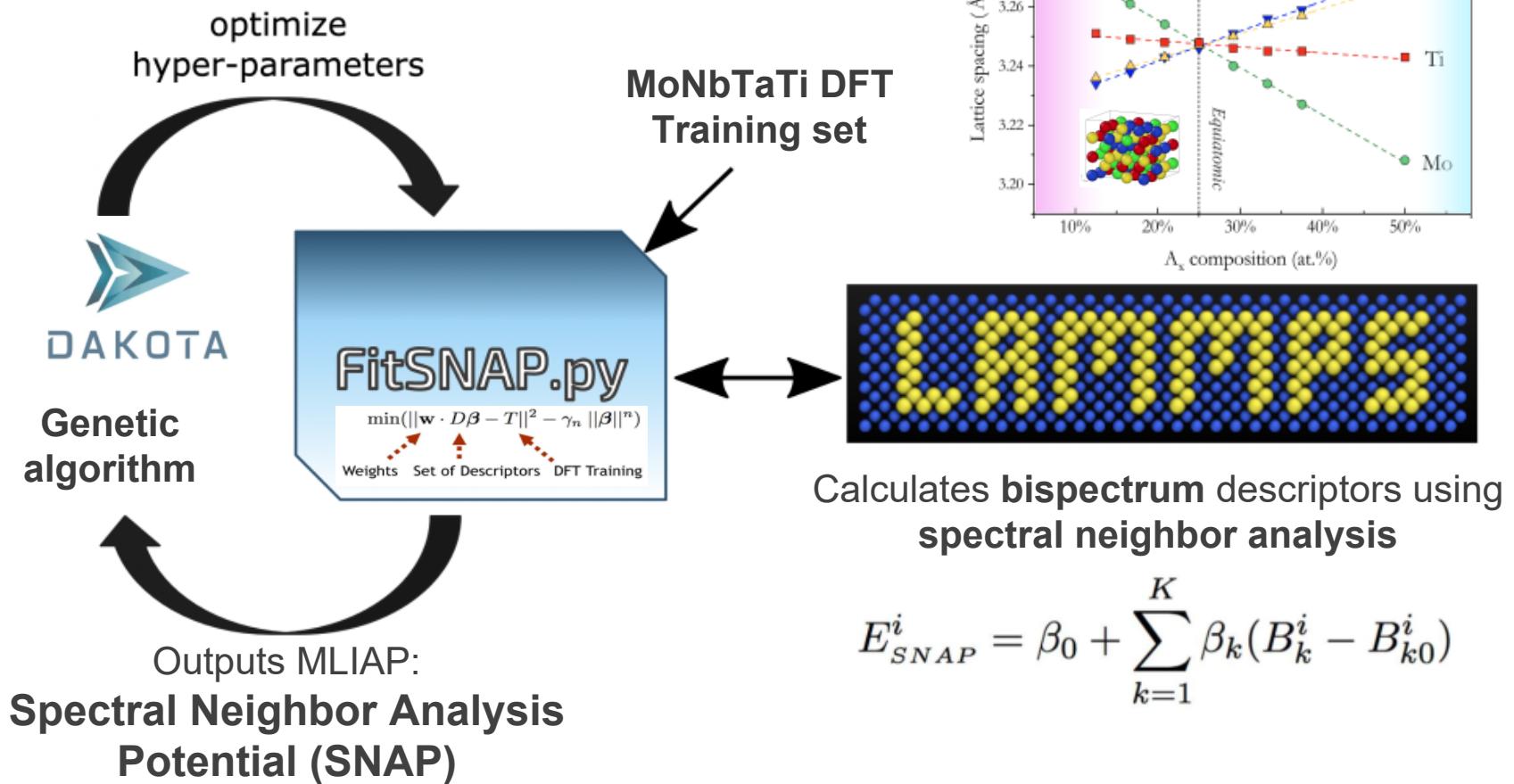
# Exploring MoNbTaTi composition space - DFT

Vary element **A**, hold elements **B**, **C**, and **D** constant



Structures	CCA (per composition)	Pure (per element)
Elastic strain	~ 1.5K - 2K	~ 100
AIMD, solid	~ 2K – 3K	~ 4K
AIMD, liquid	~ 1K – 2K	~ 3K
Volumetric strain	~10 – 20	~ 25
Uniaxial strain	~15 – 20	-
Surfaces	~ 10 – 20	~ 50 – 75

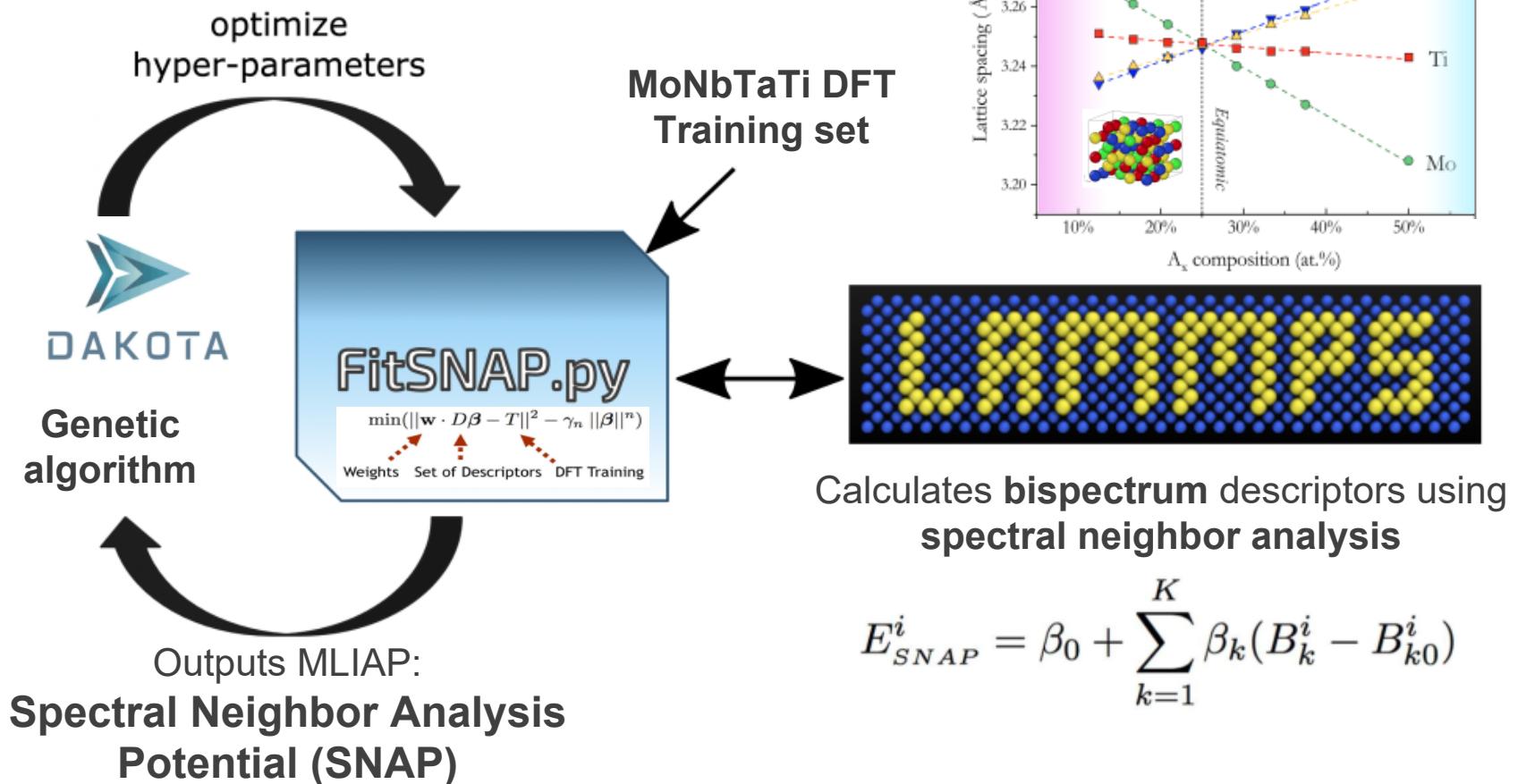
# 9 MLIAP training scheme



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**Goal 1:** Understand 'yield' of rich composition-varied training set (no additional structures)

→ MLIAP are trained to match **trends in elastic properties with changes in composition**

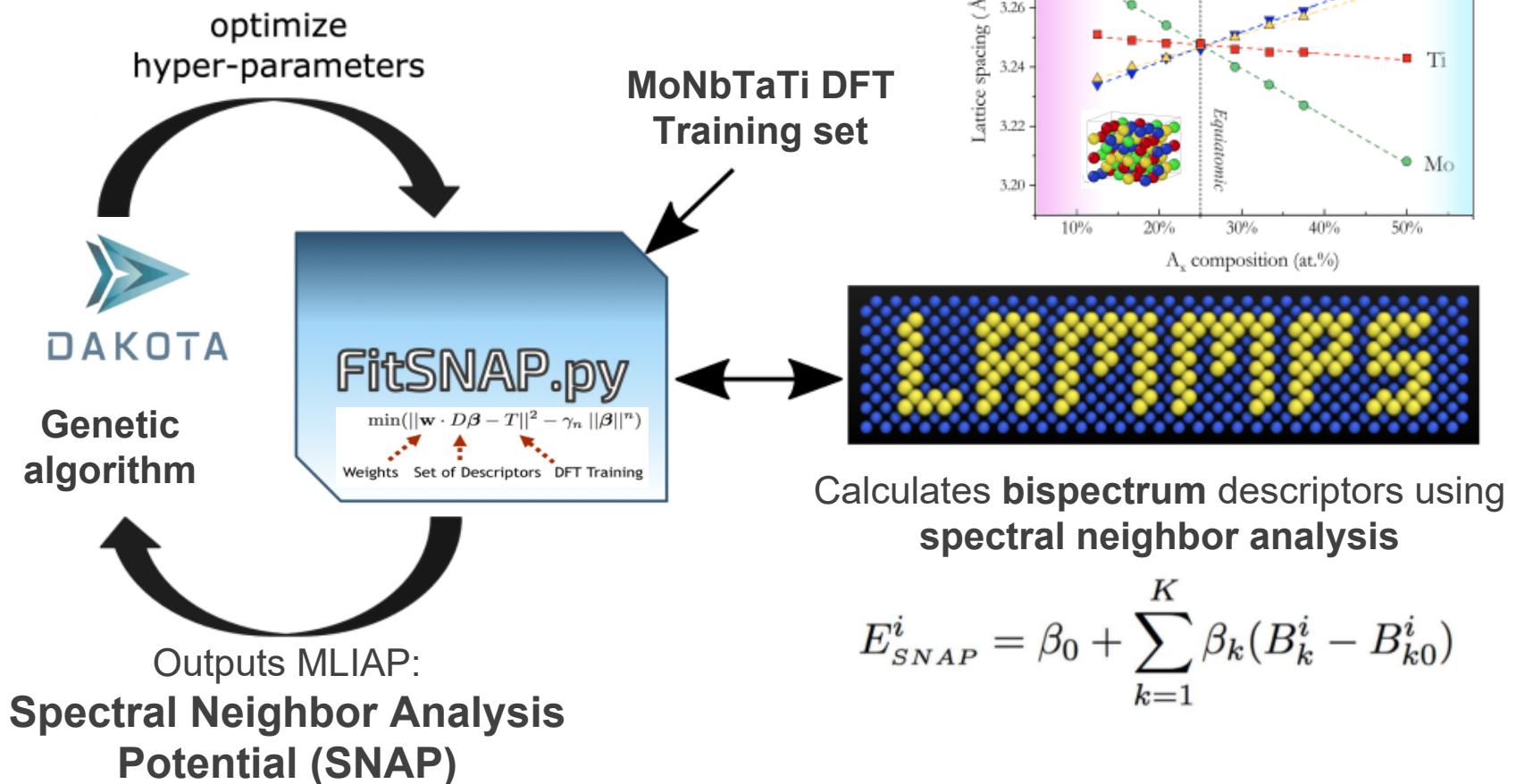


$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_{k0}^i)$$

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→ MLIAP are trained to match **trends in elastic properties with changes in composition**



$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_{k0}^i)$$

$$E = \frac{9GB}{3B+G}$$



$$B_0 = \frac{C_{1111} + 2C_{1122}}{3}$$

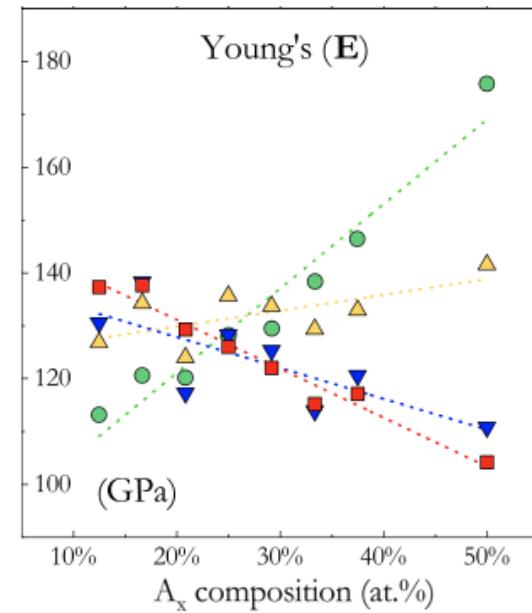
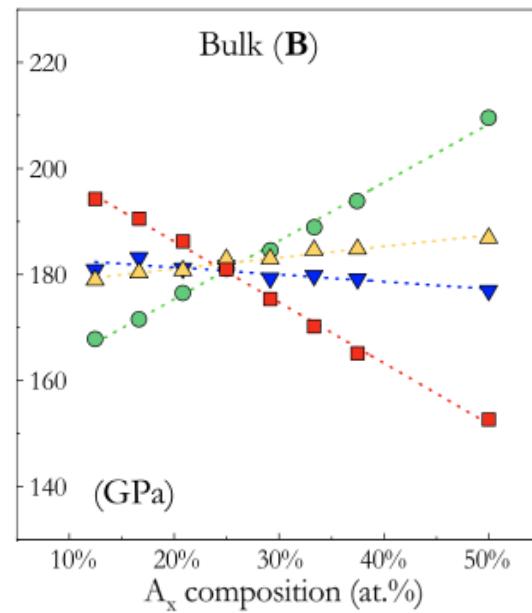
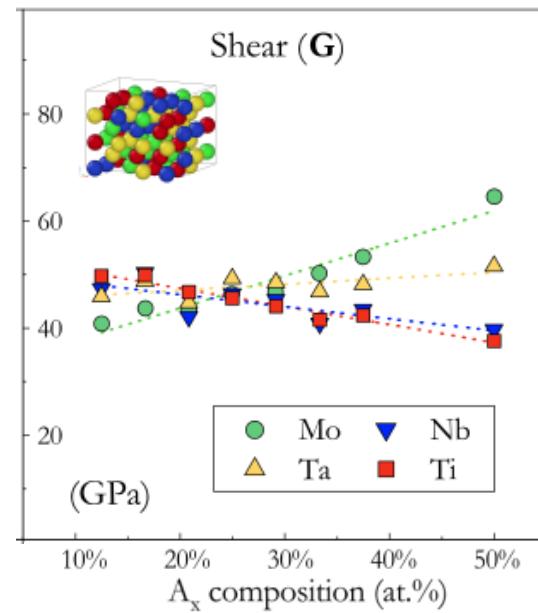
$$G = \frac{1}{2} \left[ \frac{C_{1111} - C_{1122} - 3C_{1212}}{5} + \left[ \frac{5C_{1212}}{4C_{1212} + 3(C_{1111} - C_{1122})} \right] \right]$$



Example: Young's modulus E → Genetic algorithm favors MLIAP with low errors on cubic  $C_{11}$ ,  $C_{12}$ ,  $C_{44}$

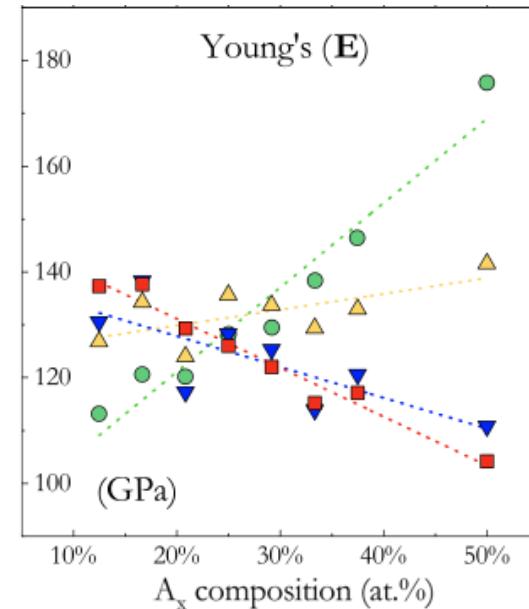
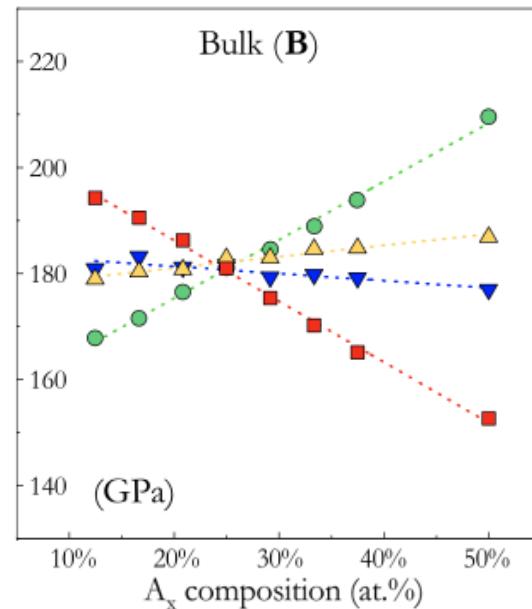
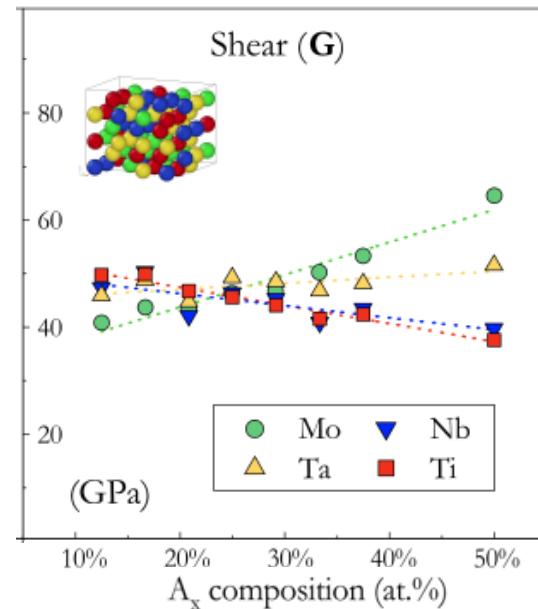
# Fits to higher-order elastic properties

DFT

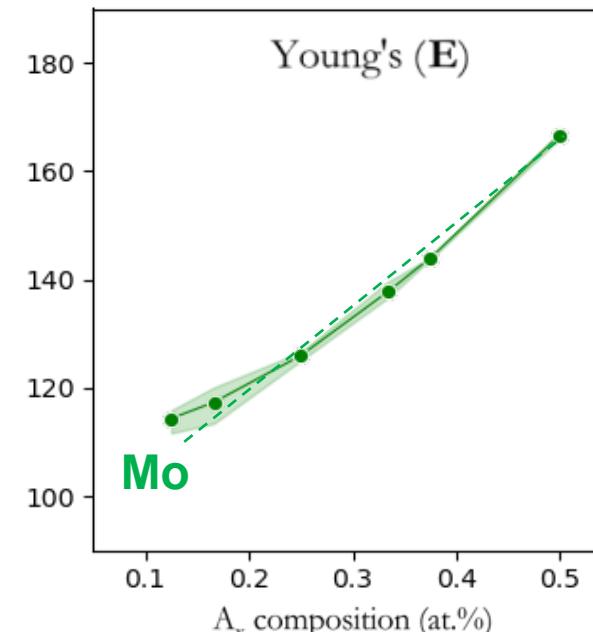
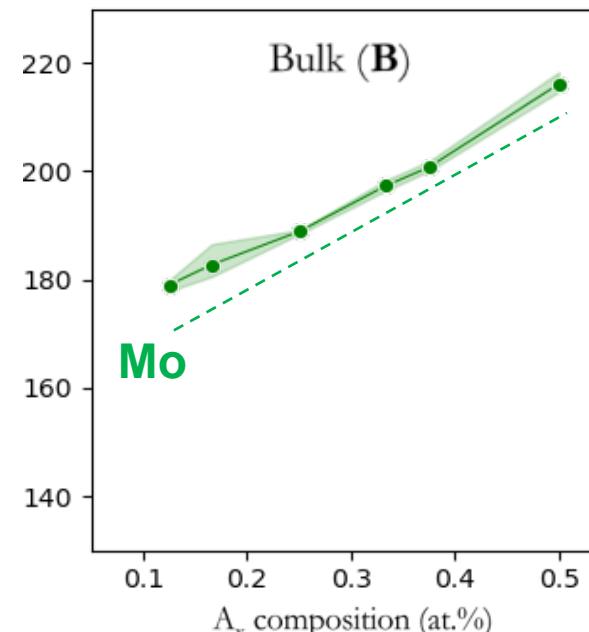
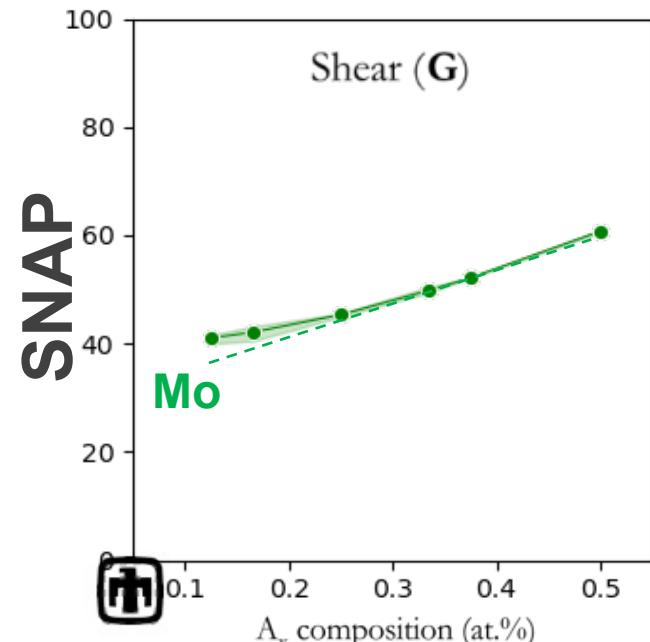


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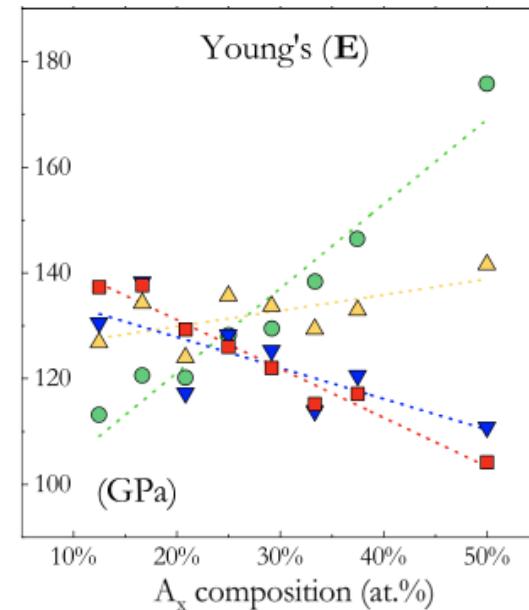
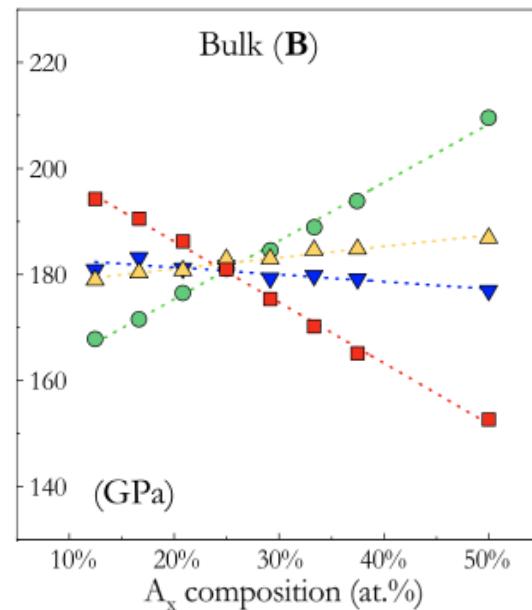
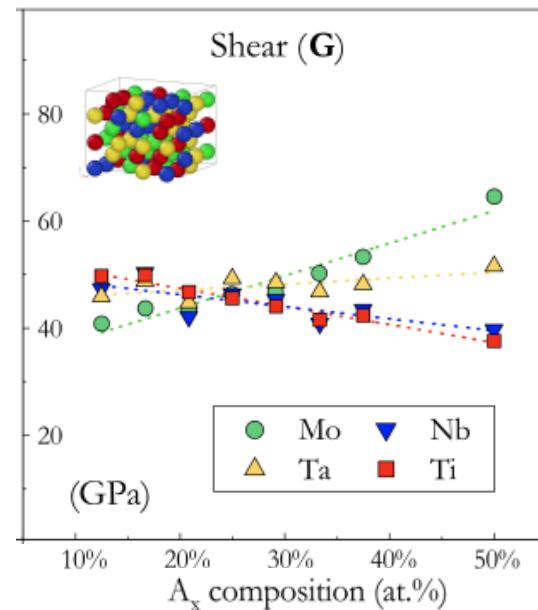


- Can reliably get stable MLIAPs that match DFT trends well across composition space for single elements
- Tested successfully at high T, P, and strain rates

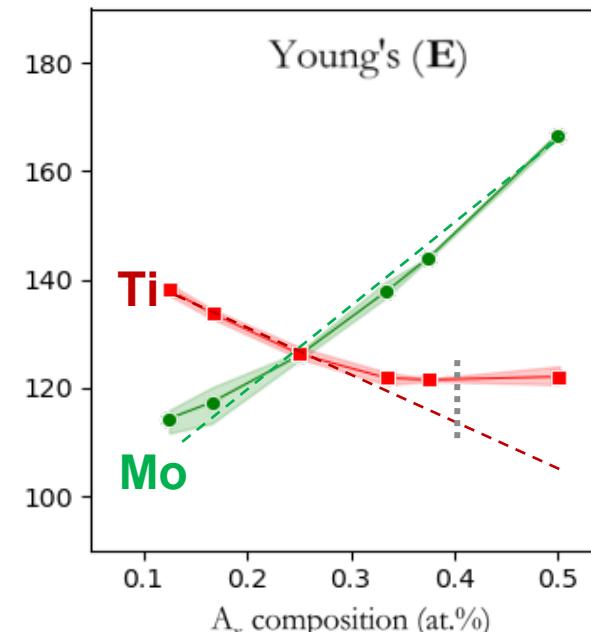
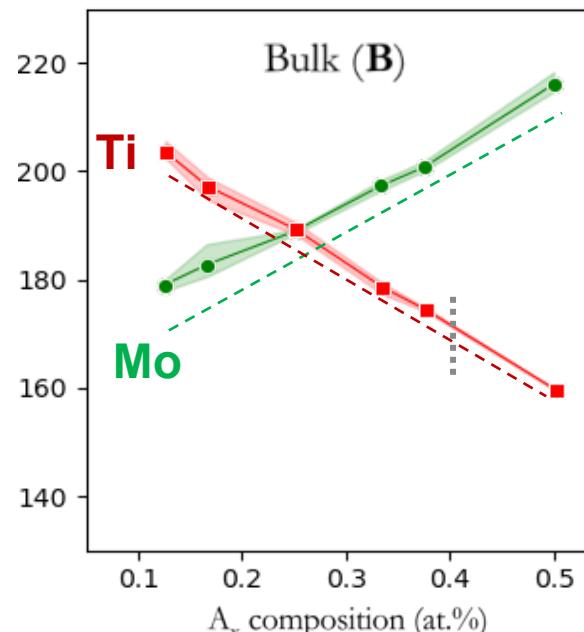
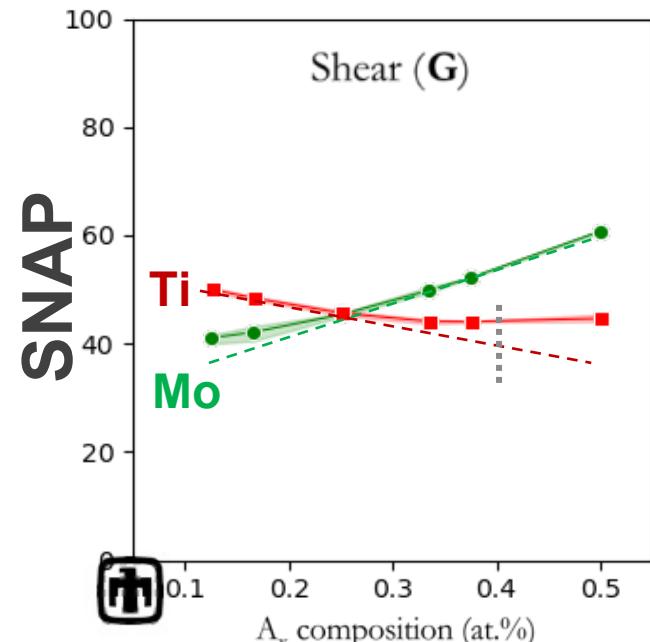


# Fits to higher-order elastic properties

DFT



- Can reliably get stable MLIAPs that match DFT trends well across composition space for single elements
- Tested successfully at high T, P, and strain rates
- Now beginning to get stable MLIAP that can match **two elements** → can extrapolate to new compositions! (here: most Mo at-%, up to Ti ~35 %)



# Extrapolation from training set

Untrained composition (vary 2 elements)



Property	SNAP 2-element MLIAPI	New DFT (not in training)	$ \text{SNAP} - \text{DFT} $
C11 (GPa)	239.6	237.6	<b>0.8 %</b>
C12 (GPa)	143.6	129.7	<b>10.7 %</b>
C44 (GPa)	39.5	37.8	<b>4.5 %</b>
B (GPa)	175.6	165.7	<b>6.0 %</b>
G (GPa)	42.7	43.6	<b>2.1 %</b>
E (GPa)	118.5	120.1	<b>1.3 %</b>

*Startt et al., Materials & Design, 213 (2022)*



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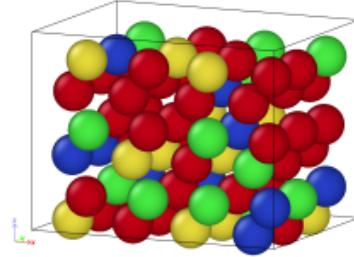
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Property	SNAP 2-element MLIAPI	New DFT (not in training)	SNAP - DFT
C11 (GPa)	239.6	237.6	0.8 %
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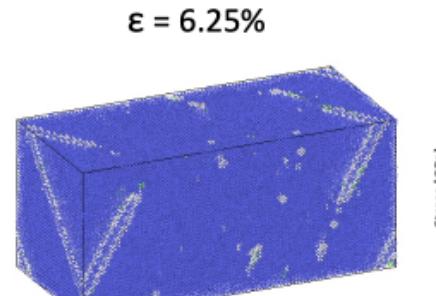
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↓ Increase system size

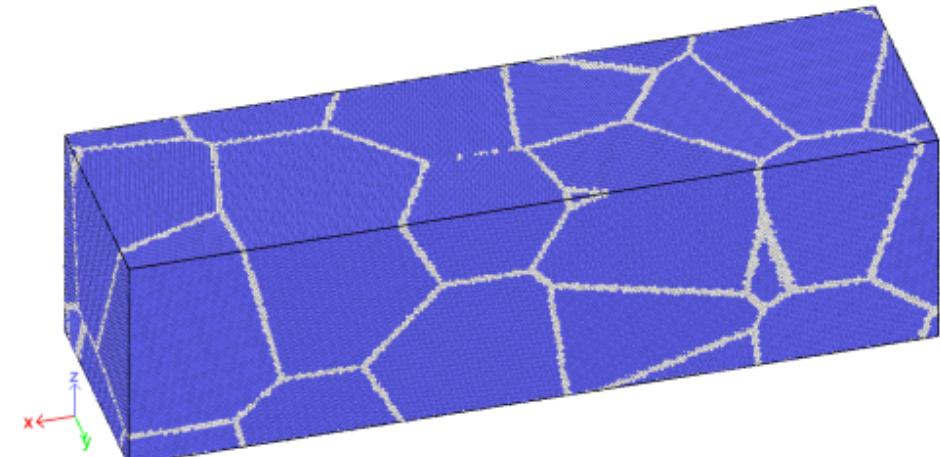
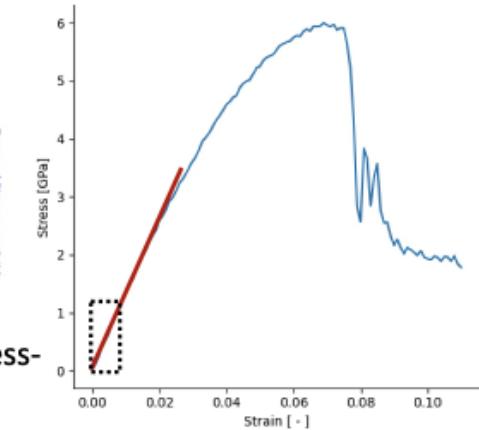


- SNAP | DFT

72-atom cell



Young's Modulus from stress-strain curve: ~120 GPa



# Extrapolation from training set

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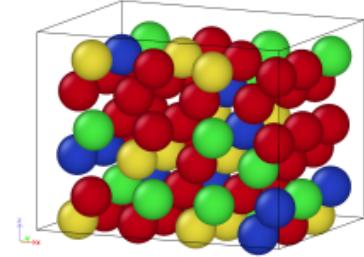
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**FitSNAP →**  
Active learning (AL) and  
uncertainty quantification (UQ)  
modules in development!

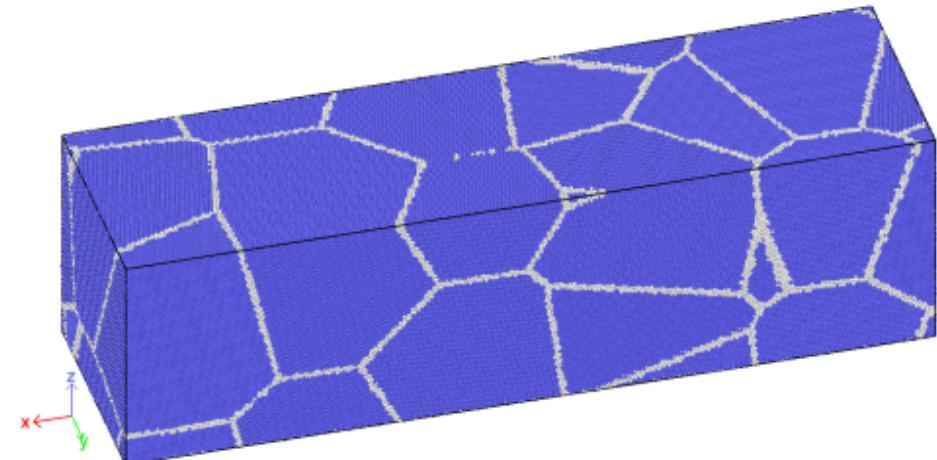
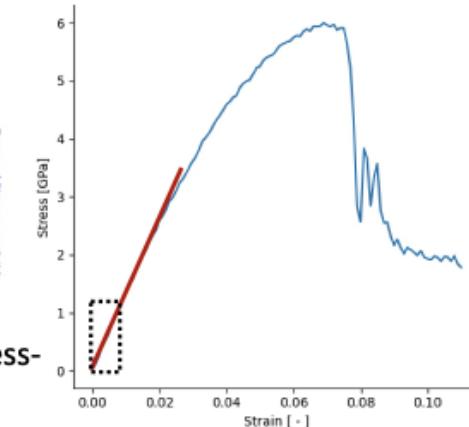
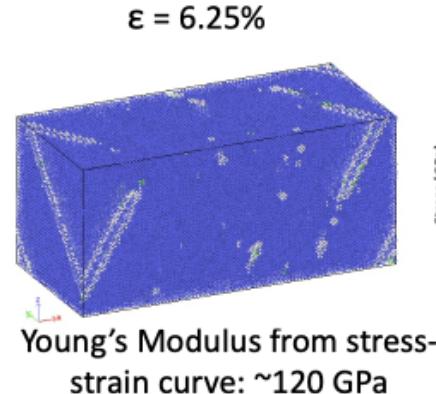


Increase system size



- SNAP | DFT

72-atom cell



Sample new MD environments, add DFT data

# Building the training set beyond elasticity



- Multiple compositions – explosion in number of training structures to include in DFT (expensive!)
  - Current MoNbTaTi comprehensive DFT set: **no point defect structures or derivative (binary, ternary) alloys**
- Use AL/UQ studies of **point defect behavior** and **alloy chemistry** to expand training set *where necessary*



# Building the training set beyond elasticity



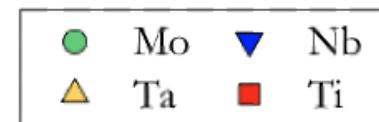
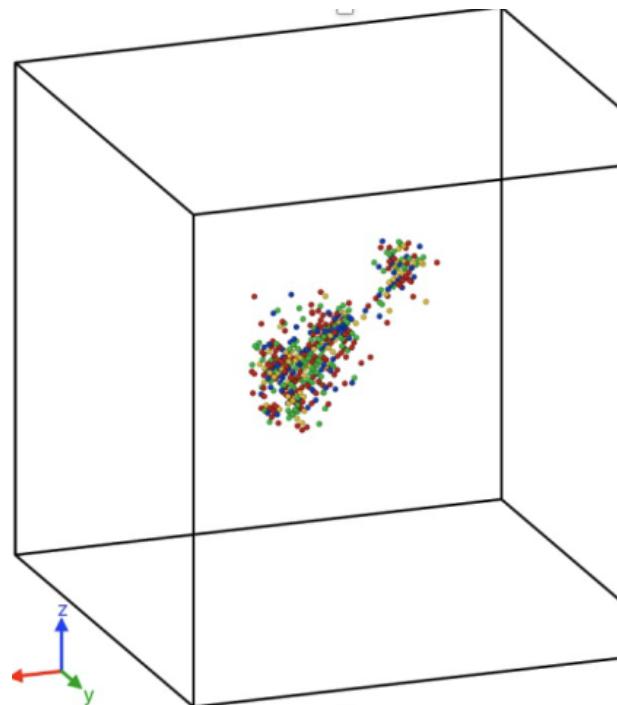
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## Point defects: Radiation damage

Primary knock-on atom (PKA) simulations →

Expose SNAP to per-atom energy levels far exceeding range seen in training set

Training: order of ~10 eV  
PKA:  
5 keV, 10 keV, 20 keV



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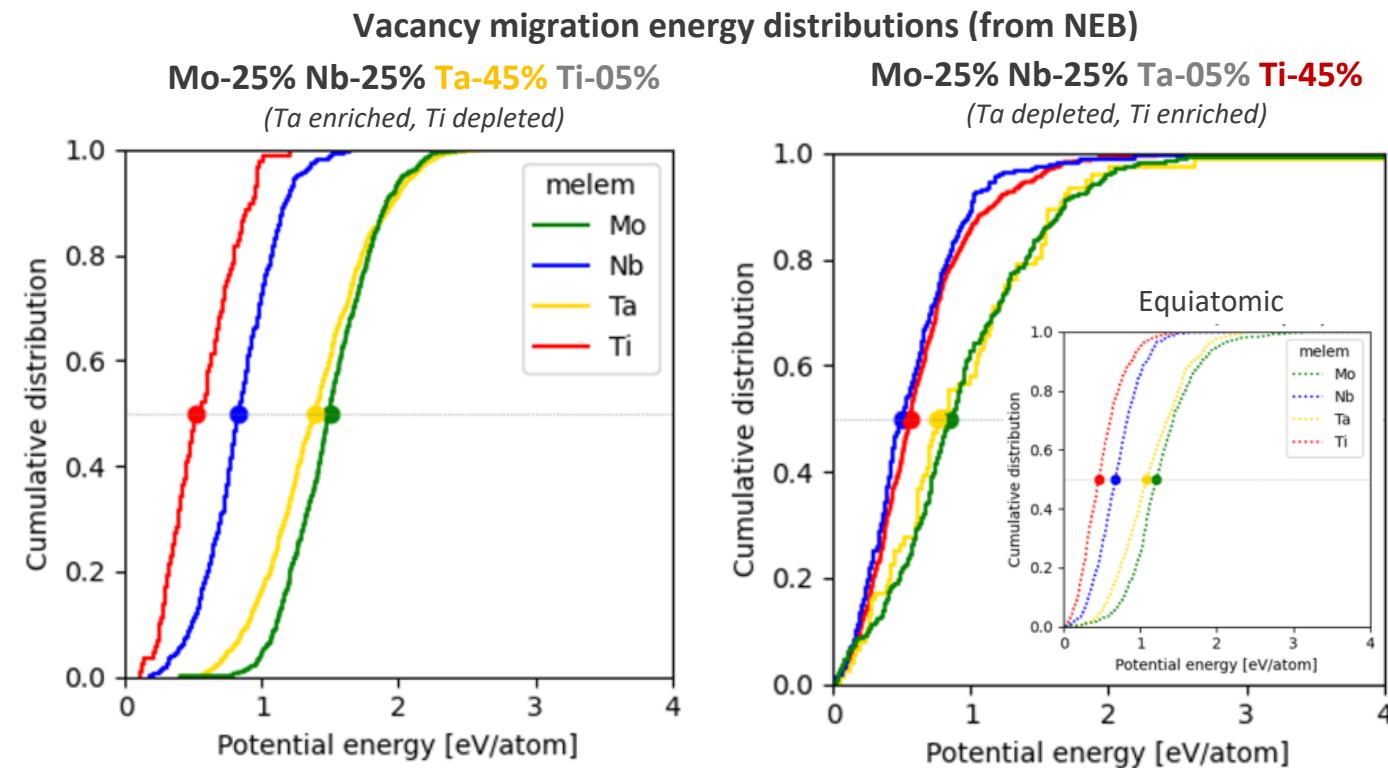
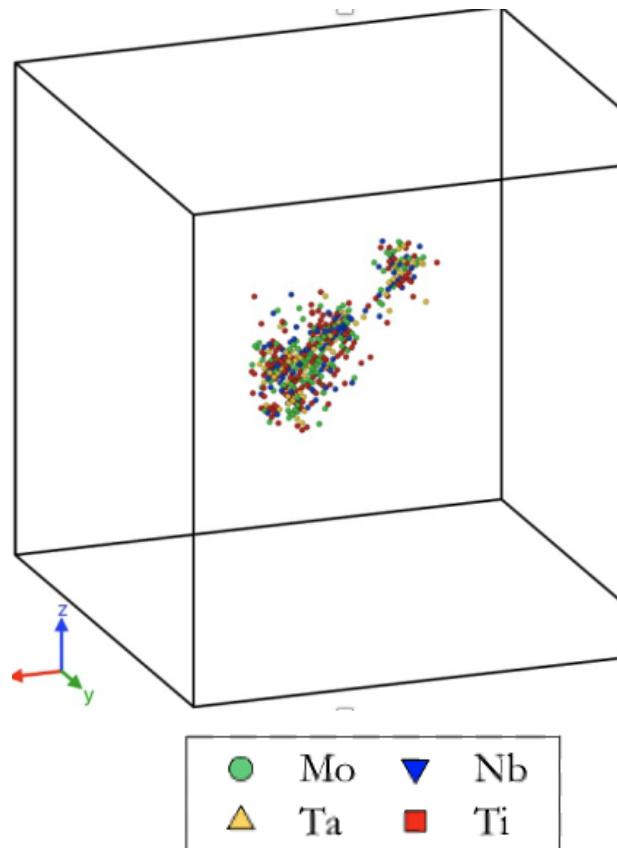


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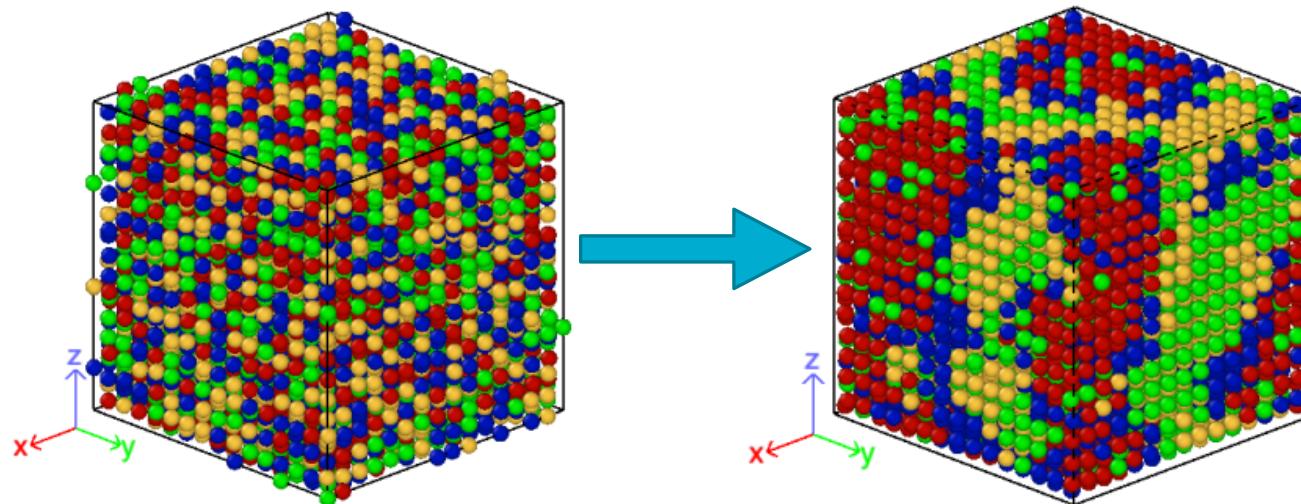


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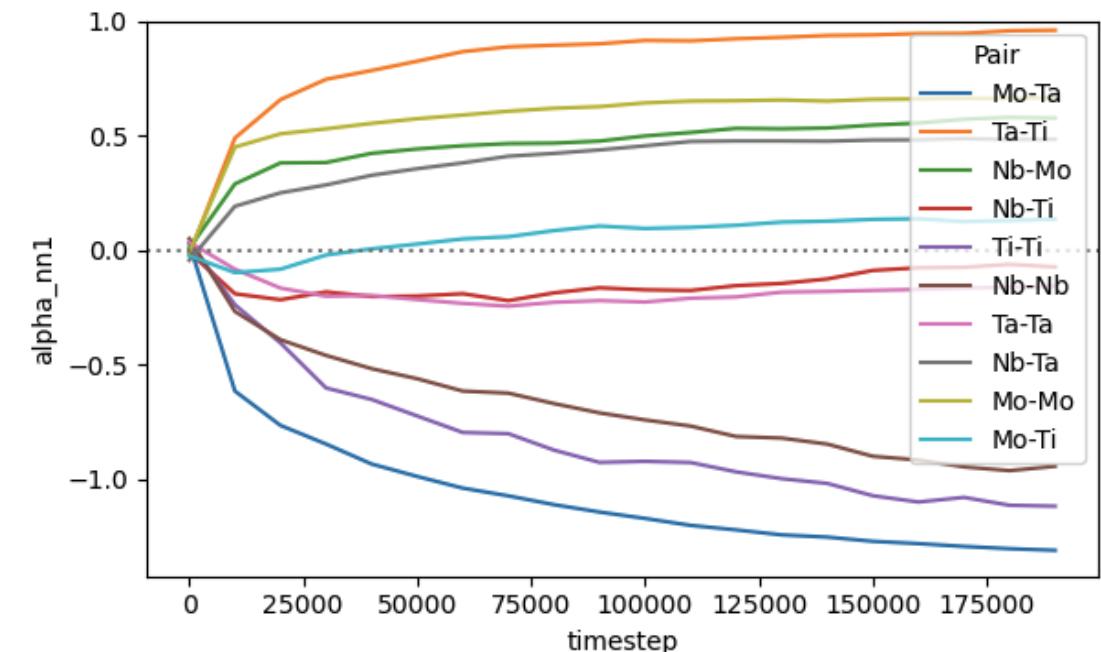
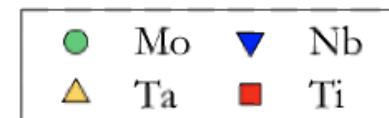


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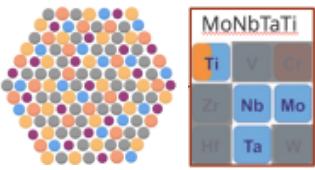
## Alloy chemistry: short-range order studies



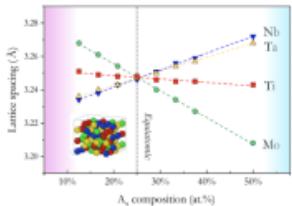
Hybrid Monte Carlo / Molecular Dynamics (MCMD) simulations →  
evolution of SRO with varying composition (also ternary)



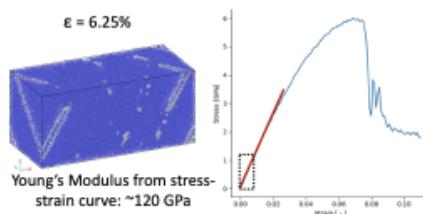
# Summary



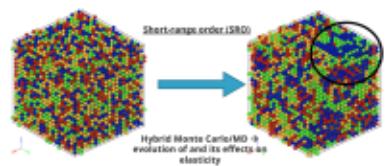
MoNbTaTi is a promising member of the refractory complex concentrated alloys



A comprehensive DFT study was undertaken by Startt et al. to understand how non-equiautomic composition affects material properties such as elasticity



Using FitSNAP and a genetic algorithm on the as-received DFT training set, we are successfully generating stable MLIAPs that can match elastic property trends across multiple compositions and extrapolate successfully to new compositions



Active learning and uncertainty quantification will be used to understand and expand scope of current MoNbTaTi training set



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