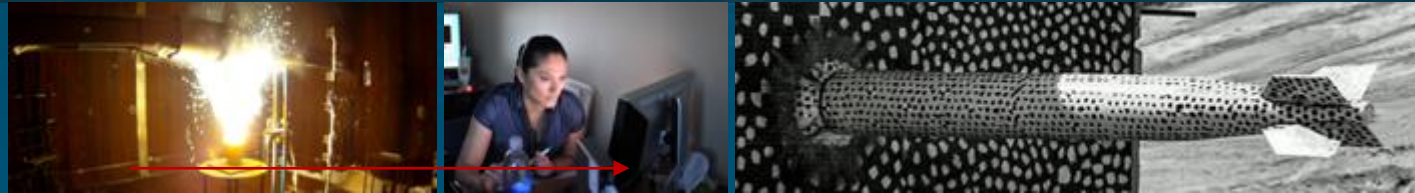




Some aspects of material stability in peridynamics



Stewart Silling

June 21, 2022
USNCTAM 2022
Austin, TX

SAND2021-0000



Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

- Working virtually created many challenges.



Dear Boss,
Can't do any work today. Cat on computer.

- But returning to the office is also creating challenges.



Dear Boss,
Can't come to work today. Cat on pants.



- **The nonlocal length scale affects all aspects of material stability.**
 - Failure kinetics: “Imaginary wave speeds” are useful.
 - Material and structural instability are related, leading to models for:
 - Kink bands in composites.
 - Self-shaping of fibers.
 - Fracture nucleation is a type of material instability.
 - Phase boundaries contain an unstable core.

Some concepts of stability (local theory)

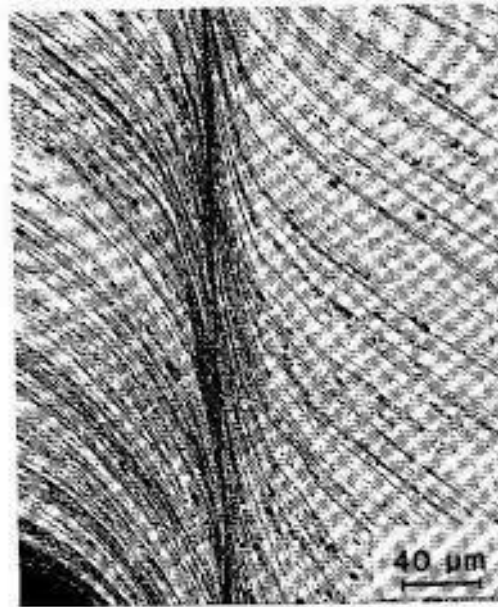
- Real wave speeds (Hadamard stability, strong ellipticity)
- Minimum potential energy
- Bifurcations
- Well-posedness
- Discontinuity in gradient (Ordinary ellipticity)



Material vs. structural stability (local theory)



- Material instability:
 - Happens at a material point, triggered by local conditions
 - Not directly related to the geometry of the body
 - Example: Adiabatic shear band
- Structural instability:
 - Happens to the entire body collectively
 - Example: Buckling of a beam



Adiabatic shear band in aluminum
image: Baxevanis et al.,
www.ima.umn.edu/materials/2008-2009/SP7.13-31.09/8186/ima.pdf



Buckling of a column
image: Klimchik
www.researchgate.net/figure/Examples-of-buckling-in-column-www-civildb-www-highline_fig11_281183936

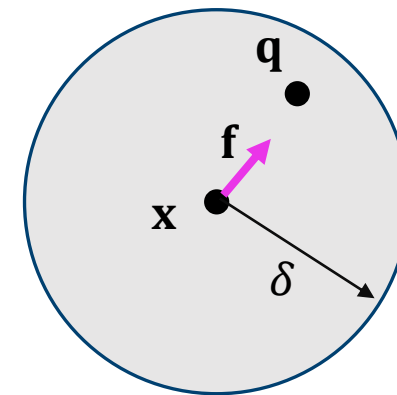
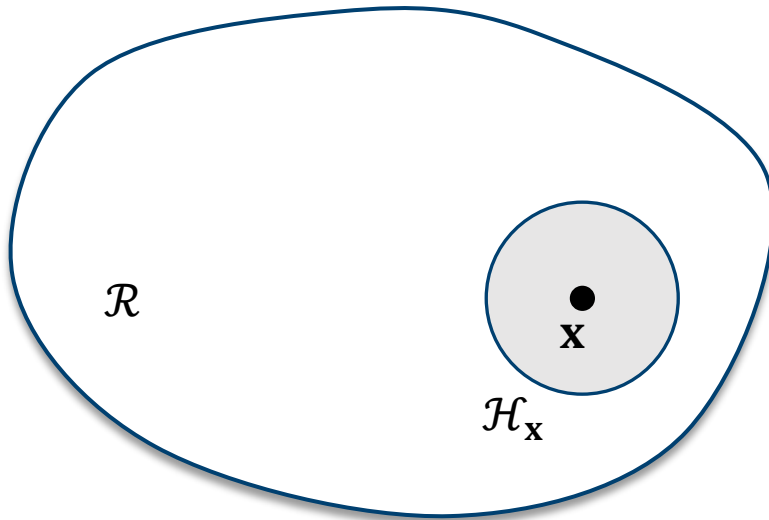
Peridynamics background



- Peridynamic momentum balance in 3D:

$$\rho(\mathbf{x})\ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}_x} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) d\mathbf{q} + \mathbf{b}(\mathbf{x}, t) \quad \forall \mathbf{x} \in \mathcal{R}, t \geq 0.$$

- \mathbf{f} is the *pairwise bond force density* of the *bond* from \mathbf{q} to \mathbf{x} .
- \mathcal{H}_x is the *family* of \mathbf{x} , which is a ball centered at \mathbf{x} with radius δ (the *horizon*).



- Later we will consider 3D deformations and bending of 1D-like structures.
- But for now, set

$$\mathbf{u} = u\mathbf{e}_1, \quad \mathbf{M} = \mathbf{e}_1, \quad \mathbf{f} = C(|q - x|)s\mathbf{e}_1$$

where C is a scalar function of the bond length called the *micromodulus* or *kernel*.

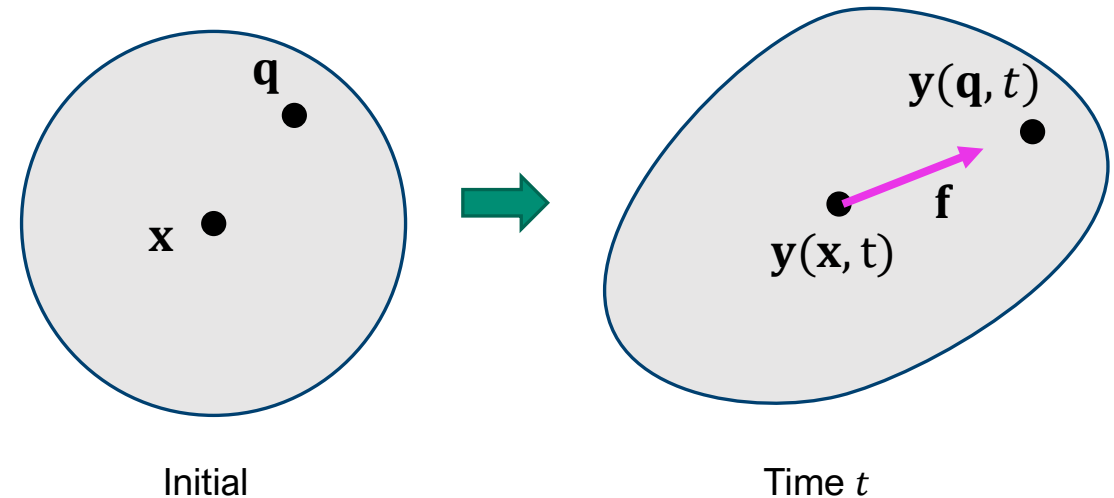
- The momentum balance is now

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi)(u(x + \xi) - u(x)) \, d\xi$$

(similar to Kunin's theory (1983)).

- C must satisfy

$$C(-\xi) = C(\xi) \quad \forall \xi.$$



An intuitive notion of stability... and a mysterious tensor



- Equilibrium:

$$\mathbf{L}(\mathbf{x}) + \mathbf{b}(\mathbf{x}) = \mathbf{0}, \quad \mathbf{L}(\mathbf{x}) = \int_{\mathcal{H}_x} \mathbf{C}(\xi)(\mathbf{u}(\mathbf{x} + \xi) - \mathbf{u}(\mathbf{x})) \, d\xi$$

where \mathbf{L} is the force density on \mathbf{x} ,

- Suppose we “carve out” a small volume surrounding a point \mathbf{x} and displace it by \mathbf{u}_0 .
- The net force density on the small volume is:

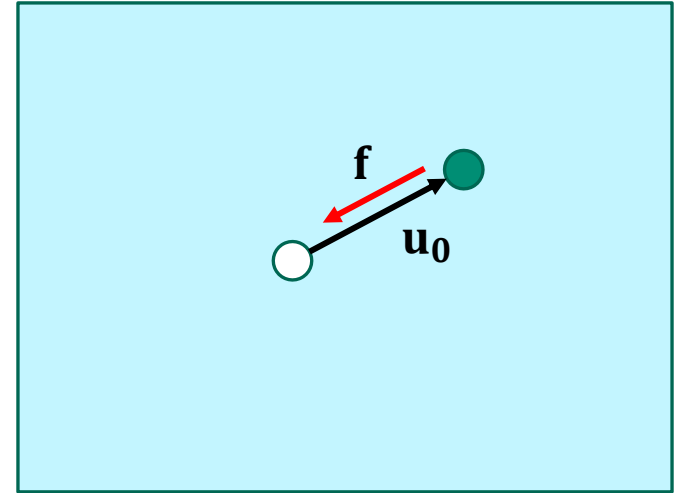
$$\mathbf{L} = -\mathbf{P}\mathbf{u}_0, \quad \mathbf{P} = \int_{\mathcal{H}_x} \mathbf{C}(\xi) \, d\xi.$$

where \mathbf{P} is the *single point response tensor*.

- *Single point stability*: If

$$\mathbf{u}_0 \cdot (\mathbf{P}\mathbf{u}_0) > 0 \quad \forall \mathbf{u}_0$$

then the particle always gets pushed back toward where it started. Otherwise \mathbf{x} can fly off to ∞ .



Comparable statement in the local theory:

$$c_{ijkl}\epsilon_{ij}\epsilon_{kl} > 0 \quad \forall \epsilon$$

Simplify further to 1D, linear microelastic

- Later we will consider 3D deformations and bending of 1D-like structures.
- But for now, set

$$\mathbf{u} = u\mathbf{e}_1, \quad \mathbf{M} = \mathbf{e}_1, \quad \mathbf{f} = C(|q - x|)s\mathbf{e}_1$$

where C is a scalar function of bond length called the *micromodulus* or *kernel*.

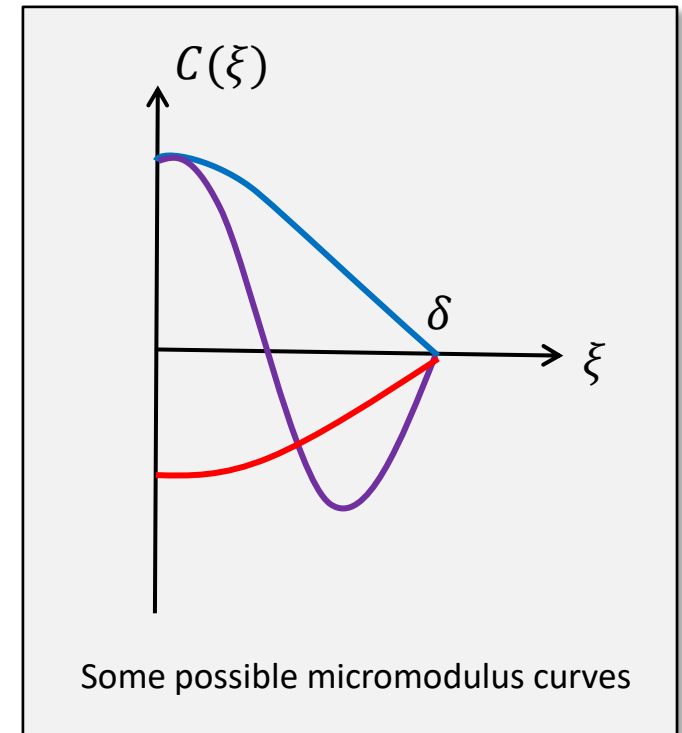
- The momentum balance is now

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi)(u(x + \xi) - u(x)) \, d\xi$$

(similar to Kunin's theory (1983)).

- C must satisfy

$$C(-\xi) = C(\xi) \quad \forall \xi.$$



Linear waves: Dispersion curves



- Assume a wave of the form

$$u(x, t) = Ae^{i(kx - \omega t)}$$

where A =amplitude, k =wavenumber, ω =frequency.

- Equation of motion:

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi) [u(x + \xi, t) - u(x, t)] d\xi$$

- leads to

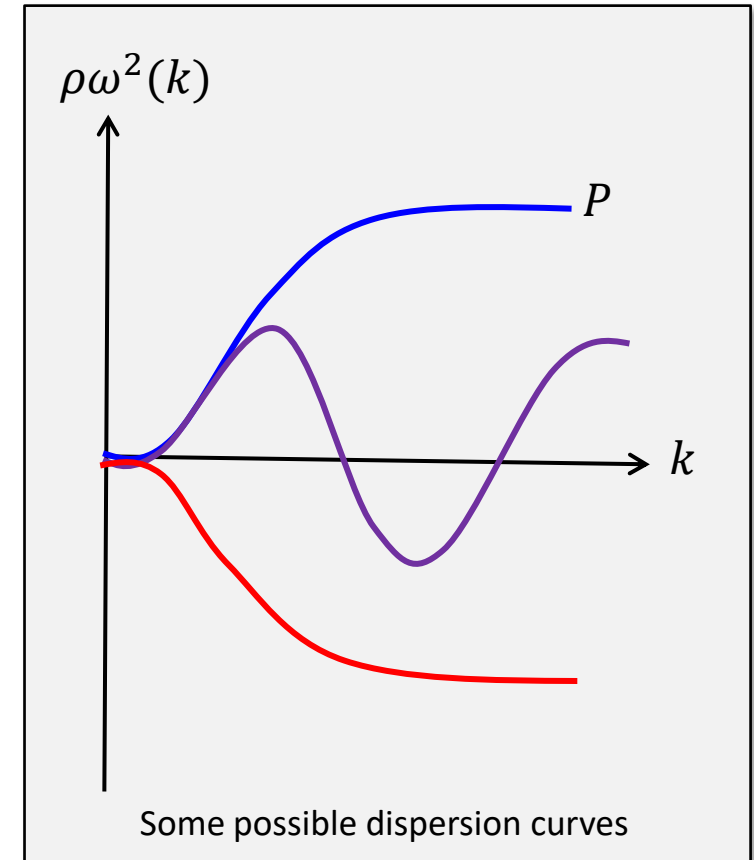
$$-\rho \omega^2 = \int_{-\delta}^{\delta} C(\xi) [e^{ik\xi} - 1] d\xi$$

- Therefore the dispersion relation is

$$\omega(k) = \sqrt{\frac{P - \bar{C}(k)}{\rho}}, \quad P = \bar{C}(0)$$

where $\bar{C}(k)$ is the Fourier transform of $C(\xi)$.

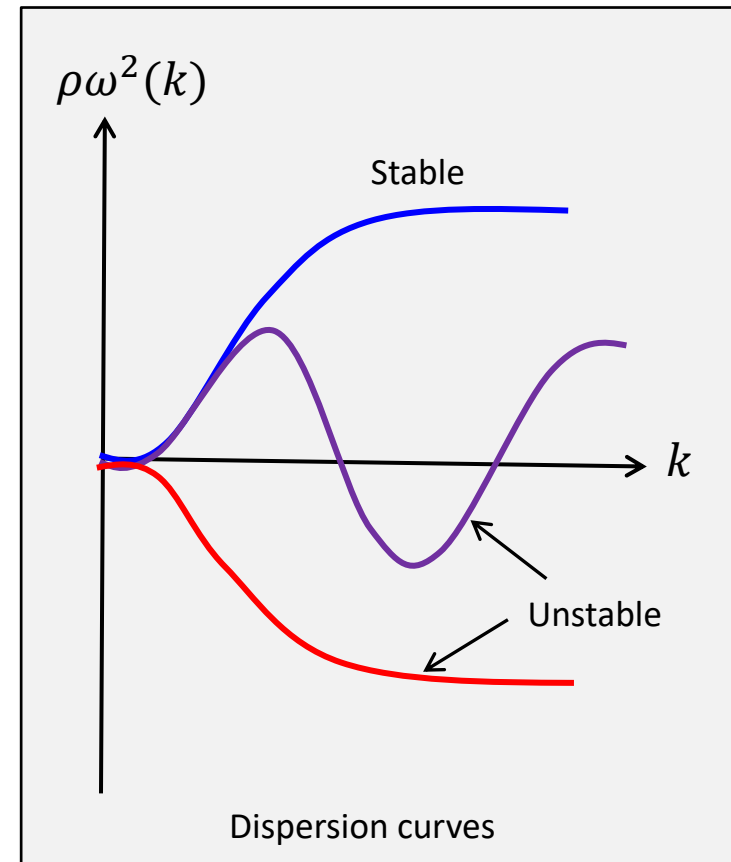
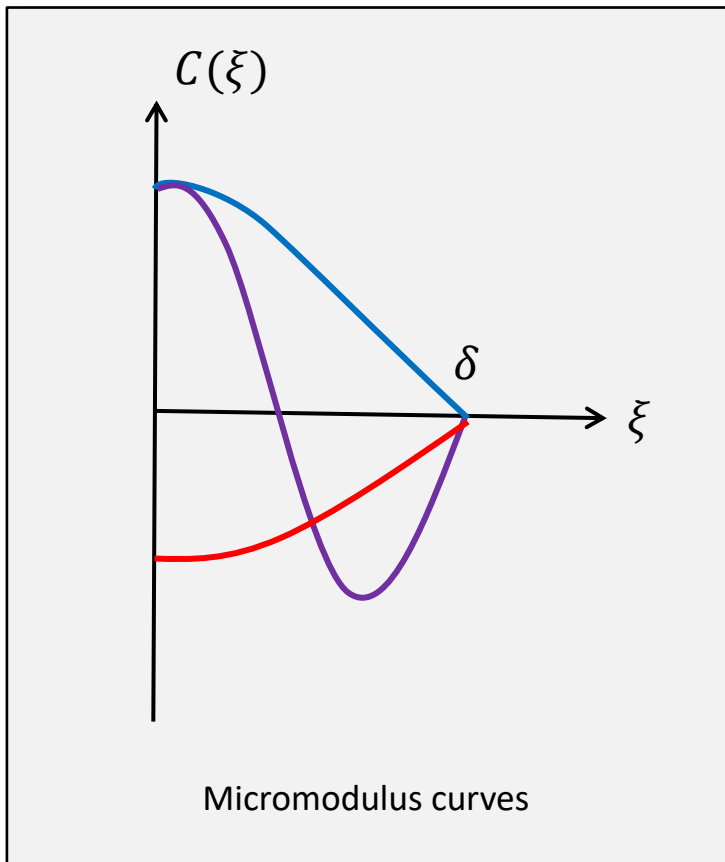
We've seen this before!



Dispersion and material stability



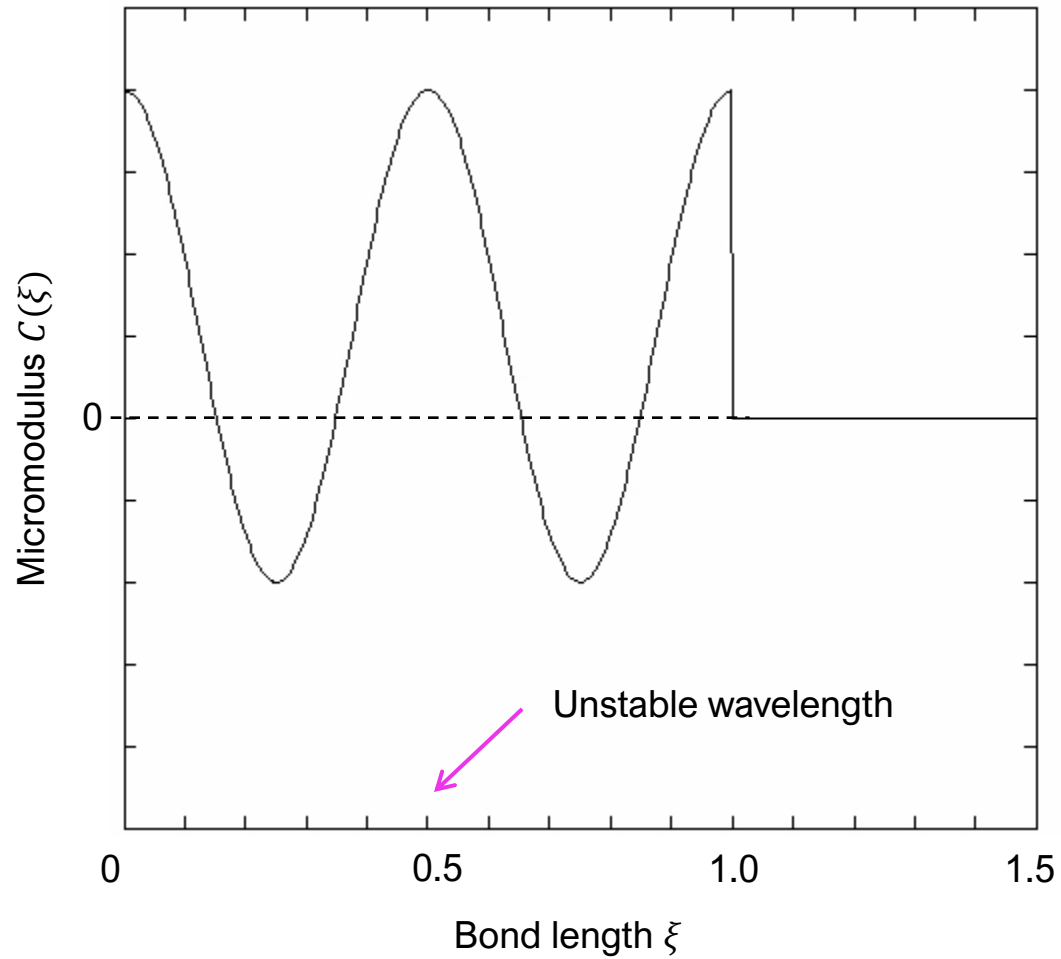
- Purple and red materials have “imaginary wave speeds” $c = \omega/k$.
 - Red: all wavenumbers are unstable.
 - Purple: only some wavenumbers are unstable.



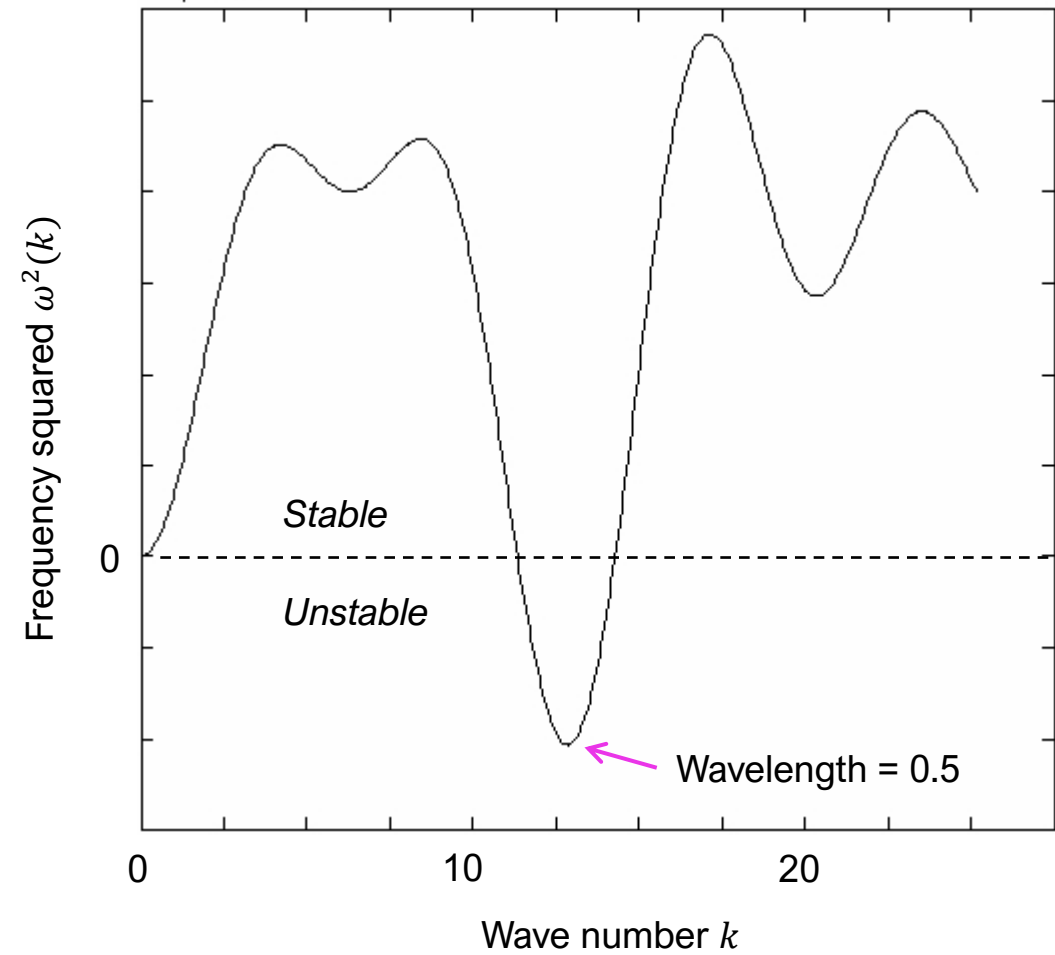
Example: A material with a narrow band of unstable wavenumbers



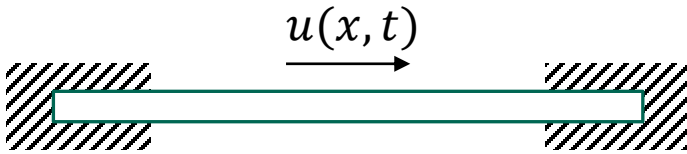
Material model



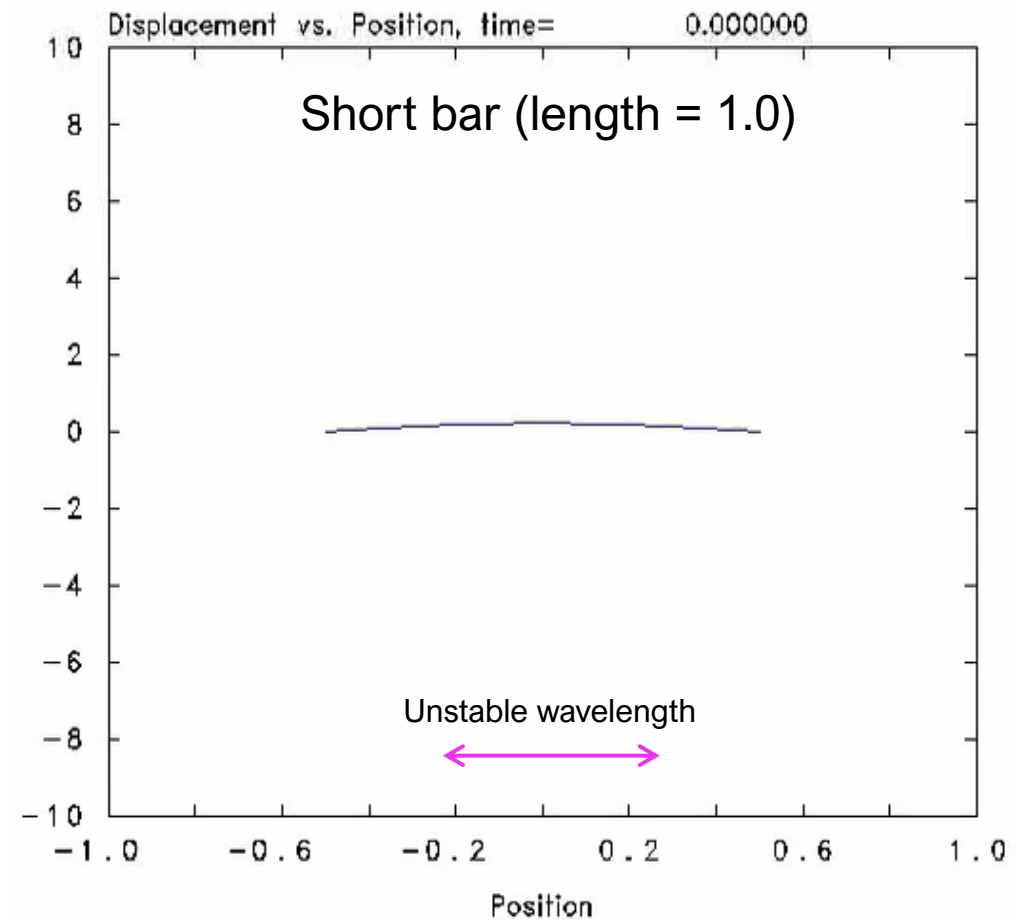
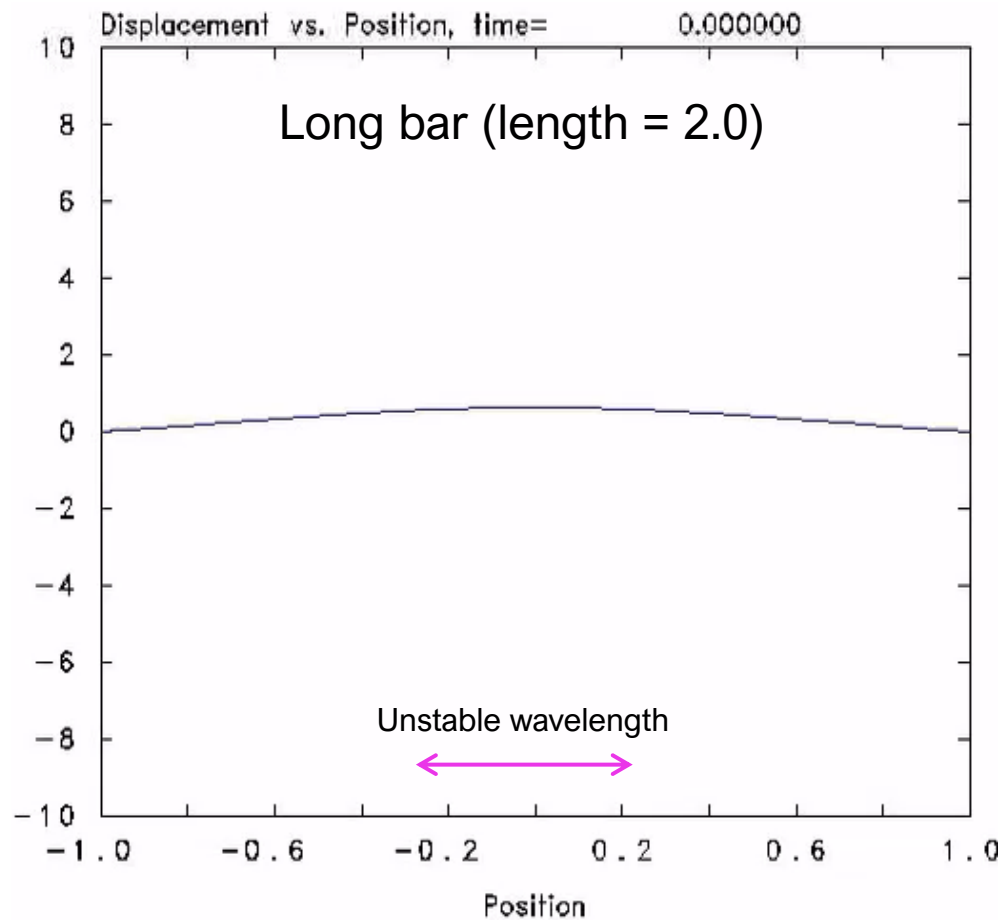
Dispersion curve



This material is stable in some geometries but unstable in others



VIDEOS



Minimum potential energy is related to wave speed



- Total potential energy in a bounded body:

$$\Phi_{\mathbf{u}} = \frac{1}{2} \int \int w(\mathbf{x}', \mathbf{x}) \, d\mathbf{x}' \, d\mathbf{x} + \int \mathbf{b} \cdot \mathbf{u} \, d\mathbf{x}$$

where w is the bond energy (*micropotential*).

- Consider a set of deformations parameterized by ε :

$$\mathbf{u} = \mathbf{u}_0 + \varepsilon \mathbf{v}$$

where \mathbf{v} is a vector field.

- Stationary Φ :

$$\frac{d\Phi}{d\varepsilon} = 0 \quad \forall \mathbf{v}$$

leads to the equilibrium equation that is satisfied by \mathbf{u}_0

$$\int \mathbf{f}(\mathbf{x}', \mathbf{x}) \, d\mathbf{x}' + \mathbf{b} = \mathbf{0} \quad \forall \mathbf{x}.$$

Minimum potential energy is related to wave speed, ctd.



- Now require Φ to be a minimum as well as stationary:

$$\frac{d^2\Phi}{d\varepsilon^2} > 0$$

for all \mathbf{v} except rigid motions.

- Leads to

$$\int \int (\mathbf{v}' - \mathbf{v}) \cdot \mathbf{C}(\boldsymbol{\xi})(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} > 0 \quad \forall \mathbf{v}$$

where

$$\mathbf{C} = \frac{\partial \mathbf{f}}{\partial \boldsymbol{\eta}}, \quad \boldsymbol{\eta} := \mathbf{u}' - \mathbf{u}.$$

Minimum energy implies real wave speeds



- Suppose Φ is minimized by \mathbf{u}_0 . Let \mathbf{v} be a standing wave (eigenmode, vibrational mode).

$$\int \mathbf{C}(\boldsymbol{\xi})(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' + \rho\omega^2 \mathbf{v} = \mathbf{0}.$$

- Multiply through by \mathbf{v} and integrate over \mathbf{x} :

$$\int \int \mathbf{v} \cdot \mathbf{C}(\boldsymbol{\xi})(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} + \rho\omega^2 \int \mathbf{v} \cdot \mathbf{v} \, d\mathbf{x} = 0.$$

- After some manipulations:

$$\int \int (\mathbf{v}' - \mathbf{v}) \cdot \mathbf{C}(\boldsymbol{\xi})(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} - 2\rho\omega^2 \int \mathbf{v} \cdot \mathbf{v} \, d\mathbf{x} = 0.$$

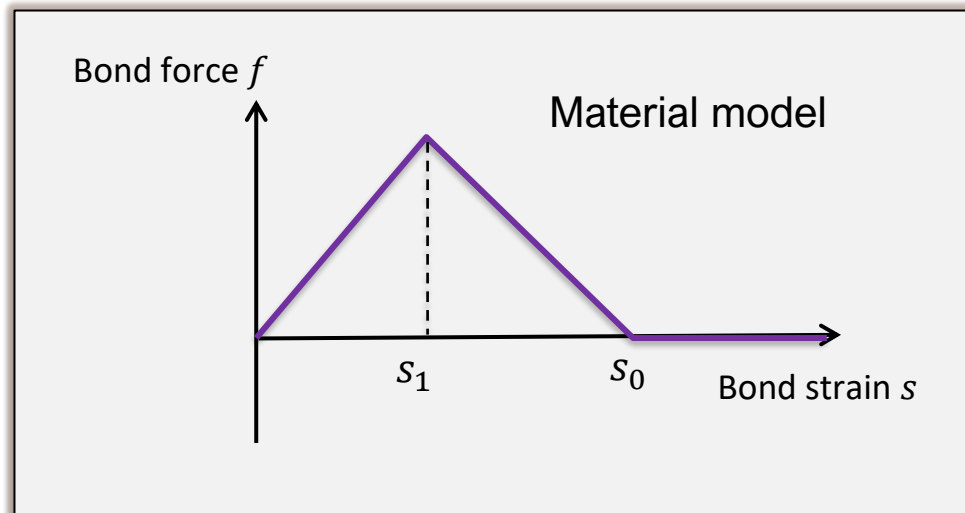
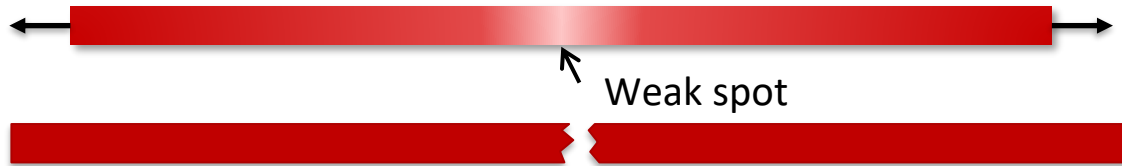
- But we already know that the $\int \int$ is positive.
- Conclude $\omega^2 > 0$. So the wave speeds are real.

Failure kinetics: How much time does it take for material to fail?

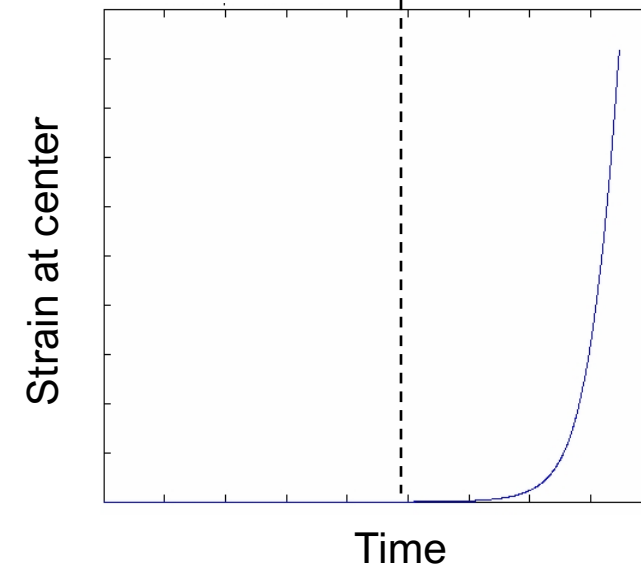
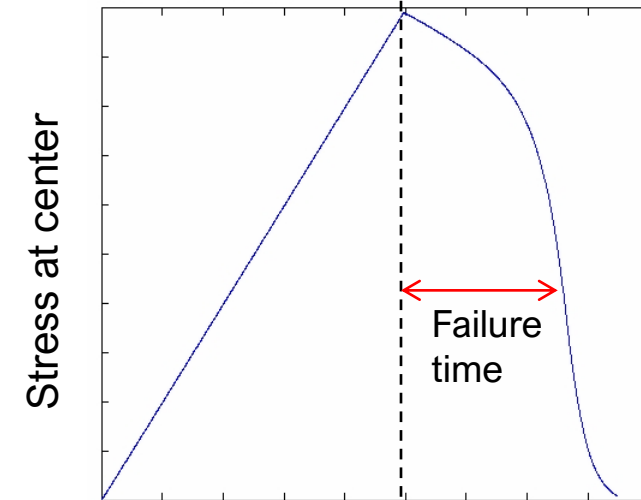


- Example: Stretching of a bar with an unstable material model.
 - Bar is stretched from the ends at a constant rate.
 - The bond force vs. strain curve has a descending branch.
 - What happens?

Strain rate



Numerical simulation results



Failure kinetics: Unstable waveforms grow exponentially but at a finite rate



- Initial data in the infinite bar:

$$u(x, 0) = A \cos kx, \quad \dot{u}(x, 0) = 0 \quad \forall x.$$

- If $\omega^2(k) > 0$:

$$u(x, t) = \frac{A}{2} [\cos(kx - \omega(k)t) + \cos(kx + \omega(k)t)].$$

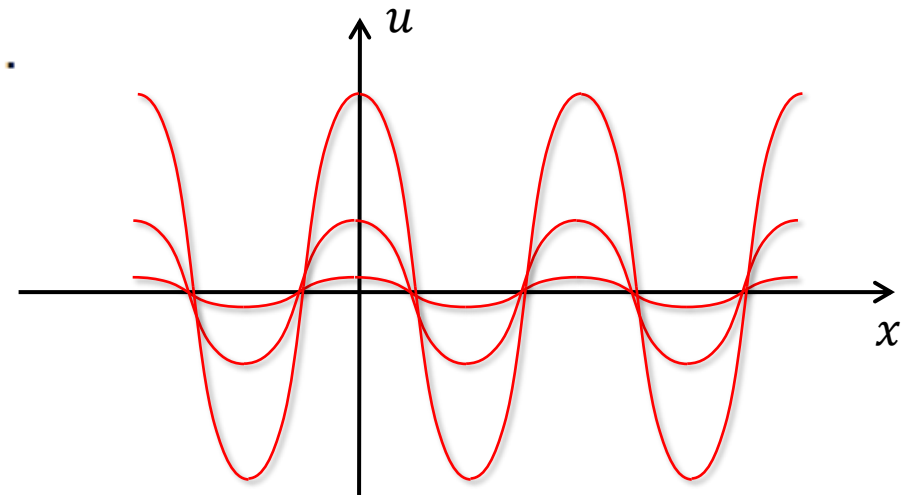
- If $\omega^2(k) < 0$:

$$u(x, t) = A \cos(kx) \cosh(\lambda(k)t)$$

where

$$\lambda(k) = \sqrt{-\omega^2} \quad \text{real.}$$

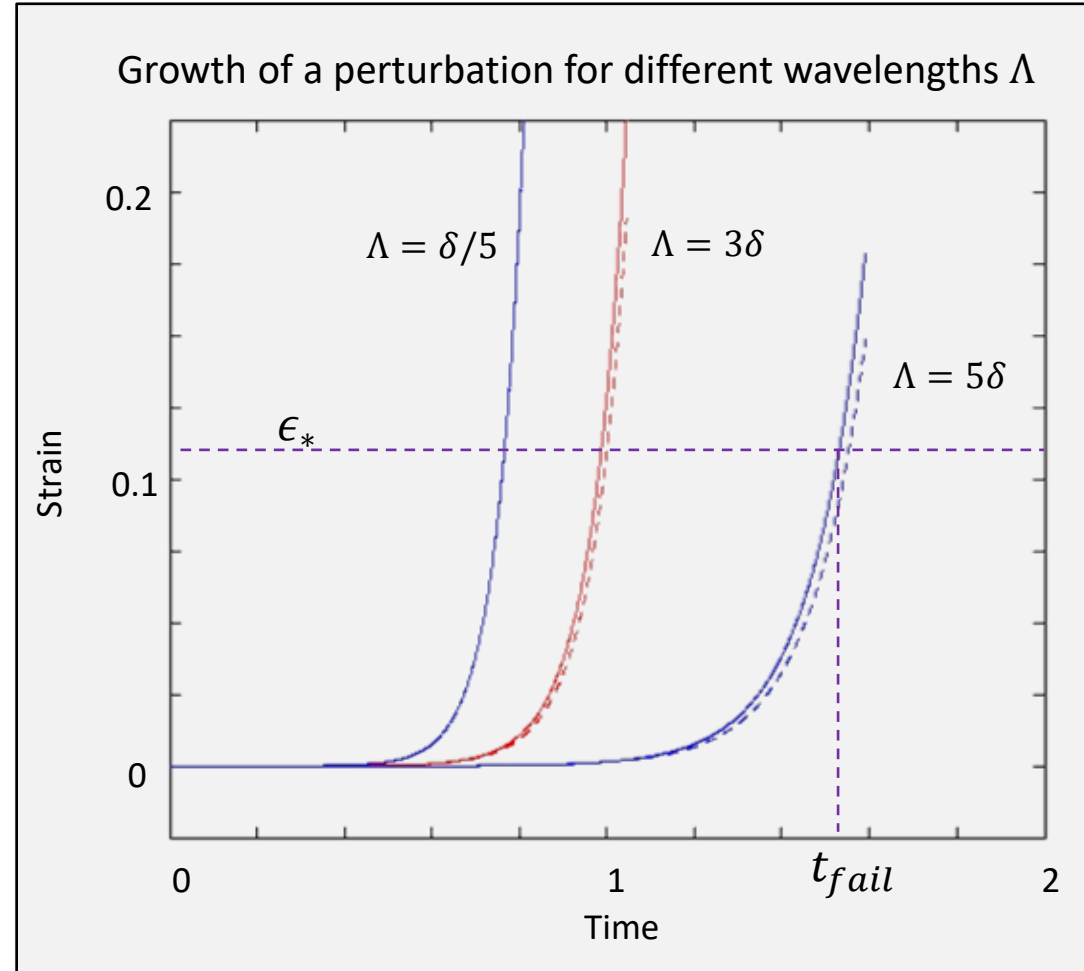
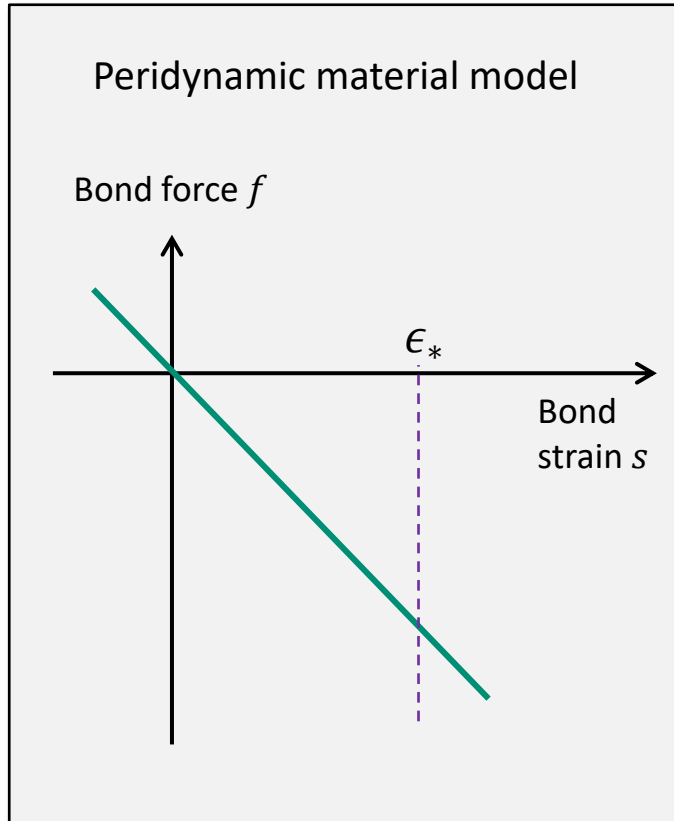
- λ is called the *blow-up rate*.



How much time does it take for an unstable waveform to grow?



- Suppose the material “fails” when the local strain u_x exceeds some given value ϵ_* .



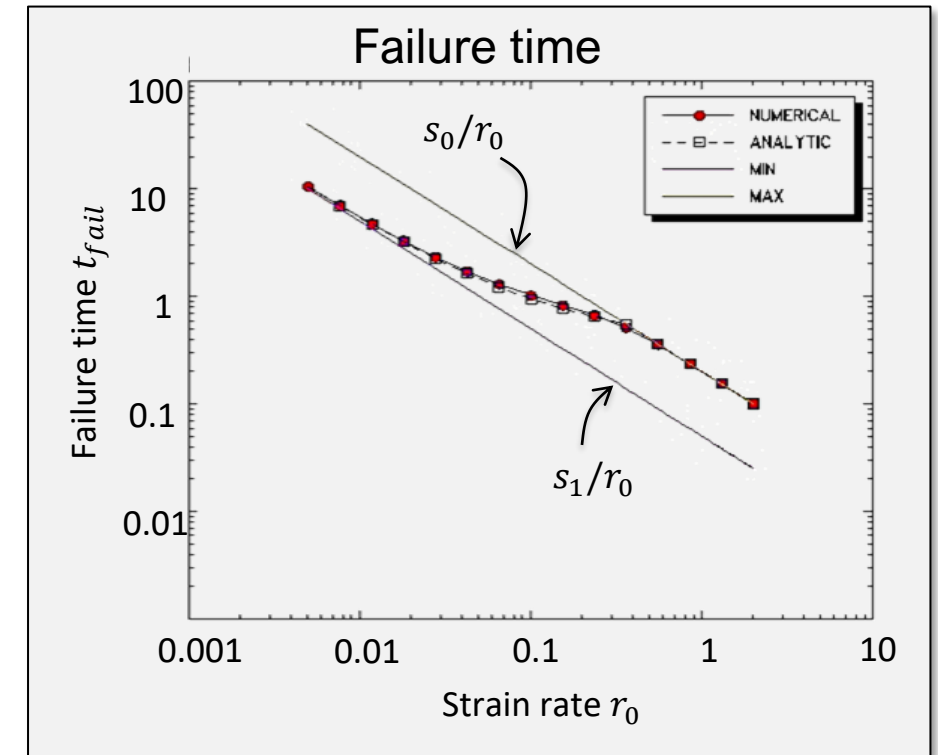
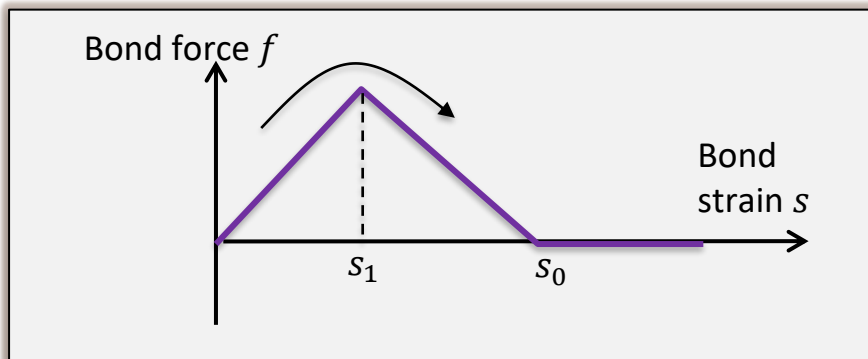
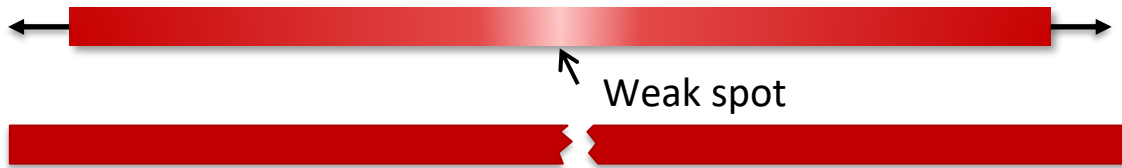
Stretching of a bar: Compute the time to failure



$$t_{fail} \approx \min \left\{ \frac{s_0}{r_0}, \left[\frac{s_1}{r_0} + \frac{1}{\lambda_\infty} \log \left(\frac{2(s_0 - s_1)}{h_0(s_1 + r_0/\lambda_\infty)} \right) \right] \right\}$$

$$\lambda_\infty = \sqrt{\frac{-P}{\rho}}$$

Strain rate



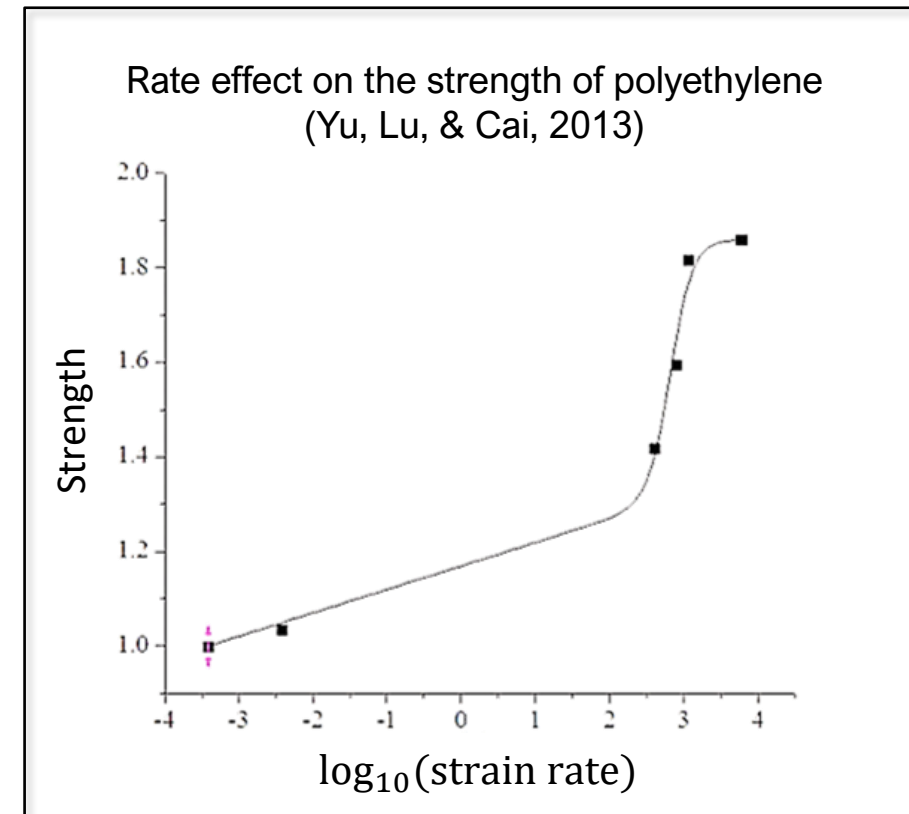
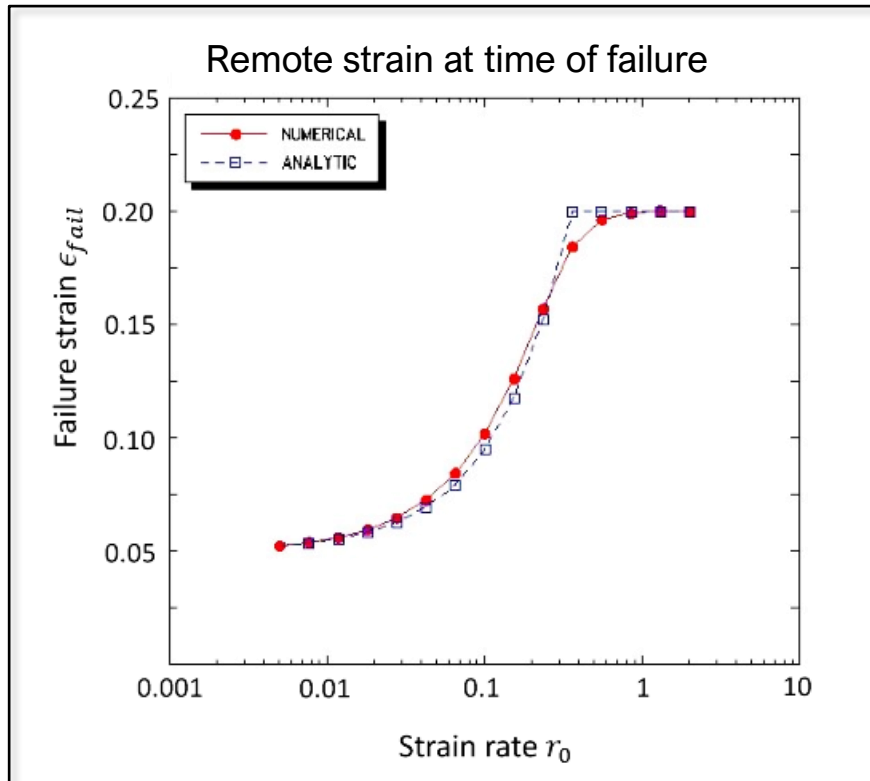
Stretching of a bar: We arrive at a macroscopic rate effect



- It takes time for the bar to fail even after some bonds have crossed the peak.
- Meanwhile, the remote strain is still increasing.
- Result is that higher strain rates lead to higher macroscopic failure strain

$$\epsilon_{fail} = r_0 t_{fail}$$

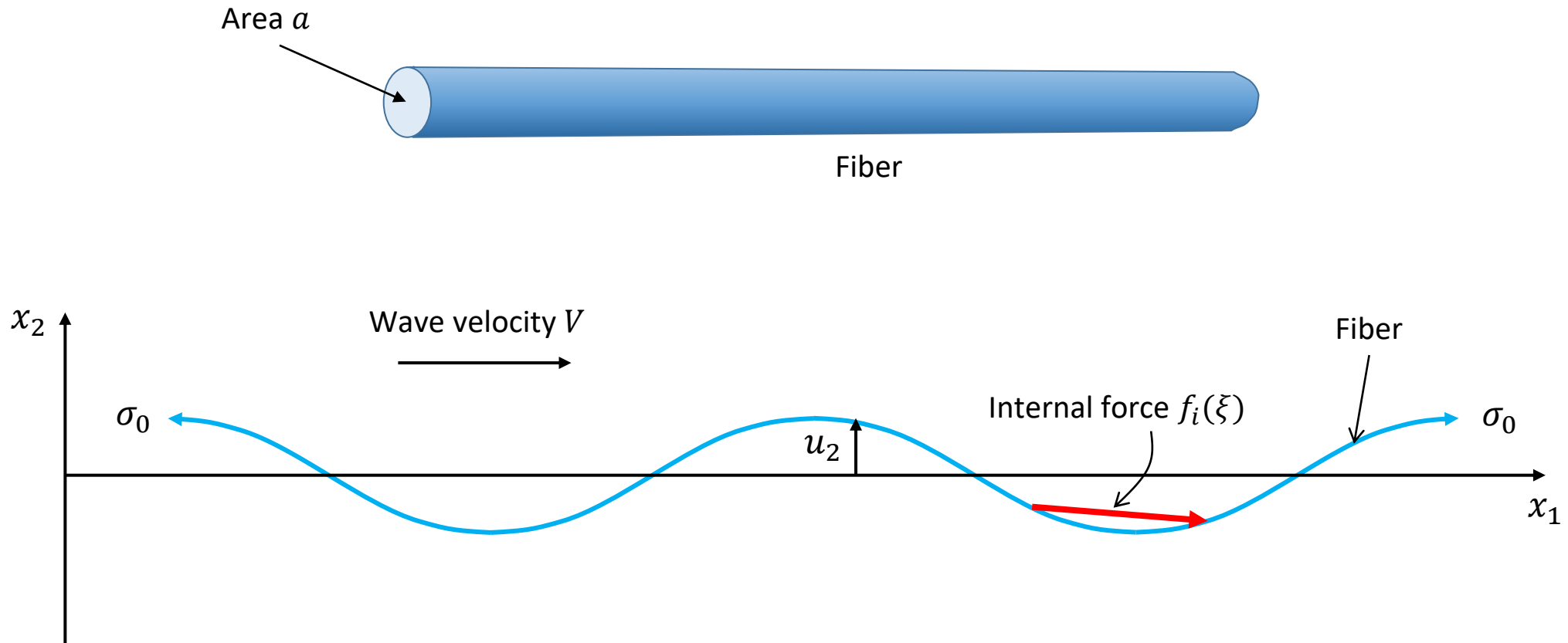
- Many real materials show a similar trend.



Instability due to internal loading in a fiber



- String made of microelastic material
- Constant long-range forces between material points
- Allow rotations (unlike true 1D)
- Study transverse waves and their stability



Internal loading in a fiber accounting for bond rotations



- Recall the 3D momentum balance:

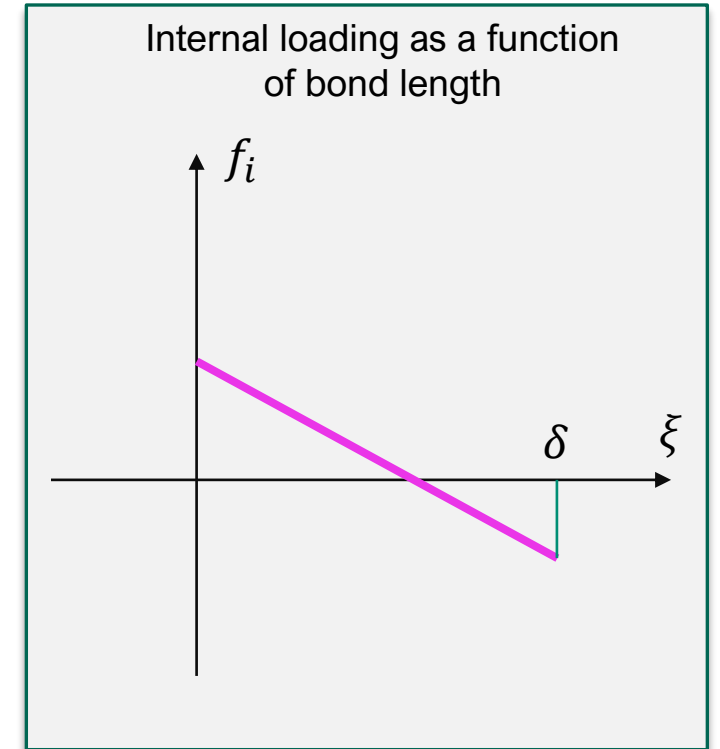
$$\rho(\mathbf{x})\ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}_{\mathbf{x}}} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) d\mathbf{q} + \mathbf{b}(\mathbf{x}, t) \quad \forall \mathbf{x} \in \mathcal{R}, t \geq 0.$$

- Allow for bond rotation.

$$\mathbf{f}(\mathbf{q}, \mathbf{x}, t) = (C(\xi)s + f_i(\xi))\mathbf{M}$$

where s is the bond strain and f_i is the prescribed internal force density in the bond ξ .

- Assume that f_i is self-equilibrated (no net axial stress).





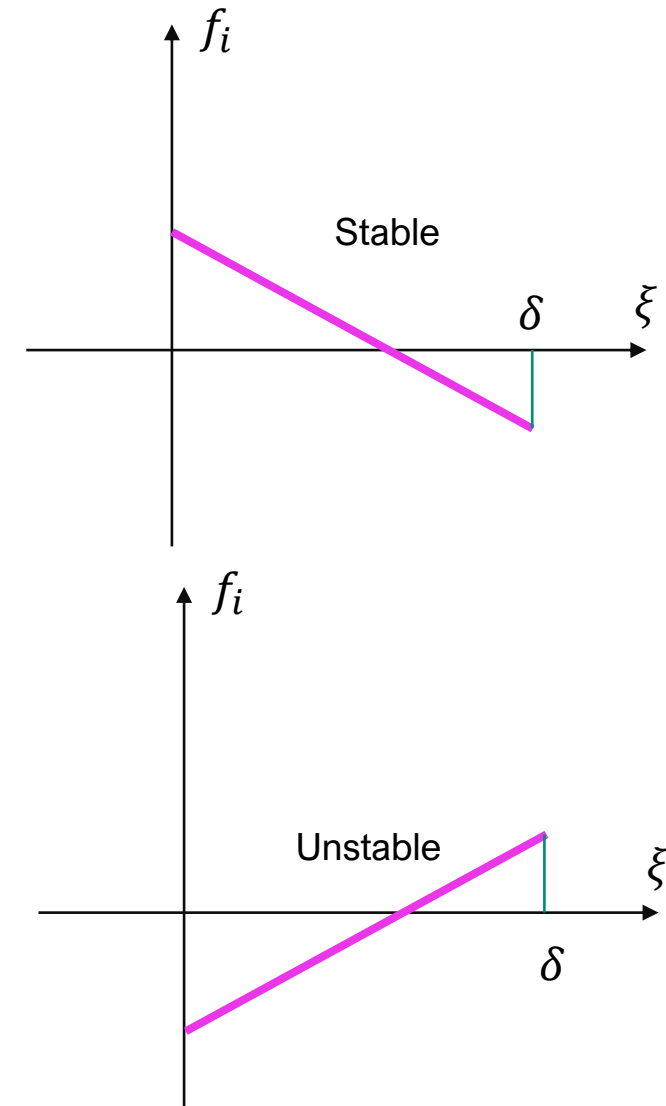
- Assume

$$u_2 = e^{i(kx - \omega t)}.$$

- Repeat derivation of the dispersion curve.

$$\rho\omega^2 = \left[\frac{-a}{12} \int_0^\delta f_i(\xi) \xi^3 d\xi \right] k^4.$$

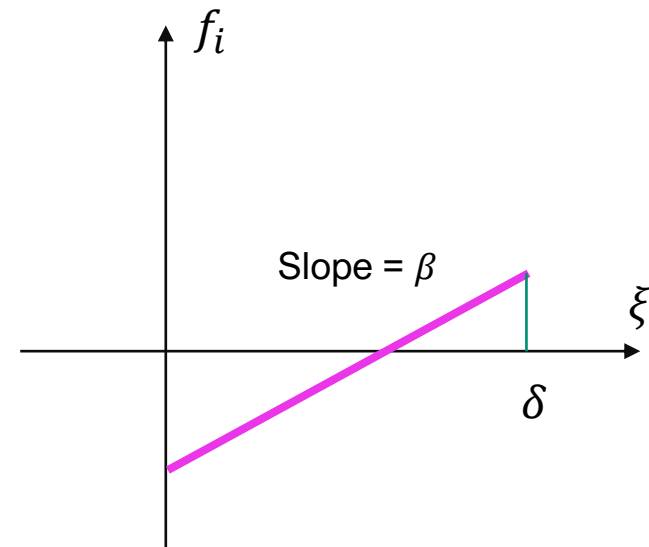
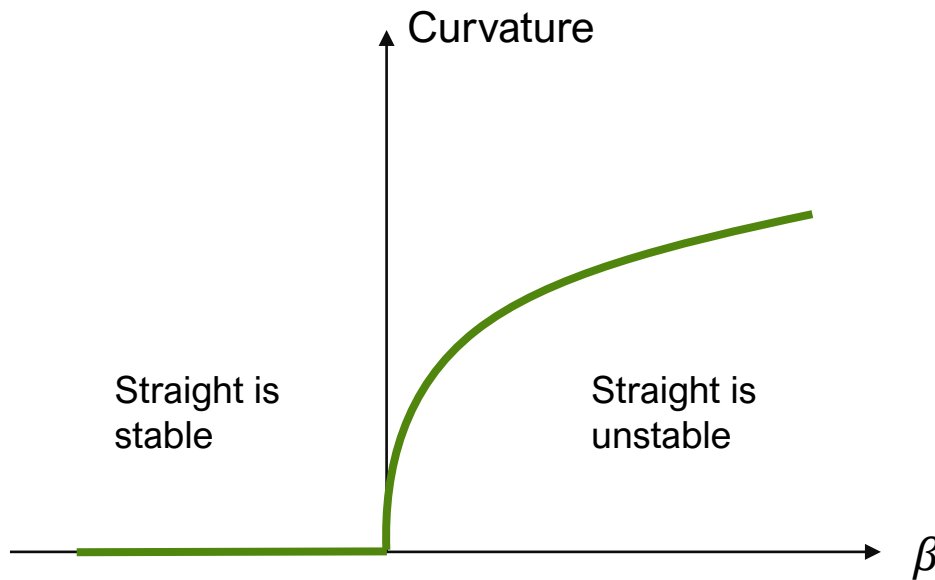
- Long bonds in compression $\implies \omega$ is real (stable).
- Long bonds in tension $\implies \omega$ is imaginary (unstable).



If a straight fiber is unstable, what does equilibrium look like?



- We can compute* the curvature of a fiber in equilibrium.
- Result:

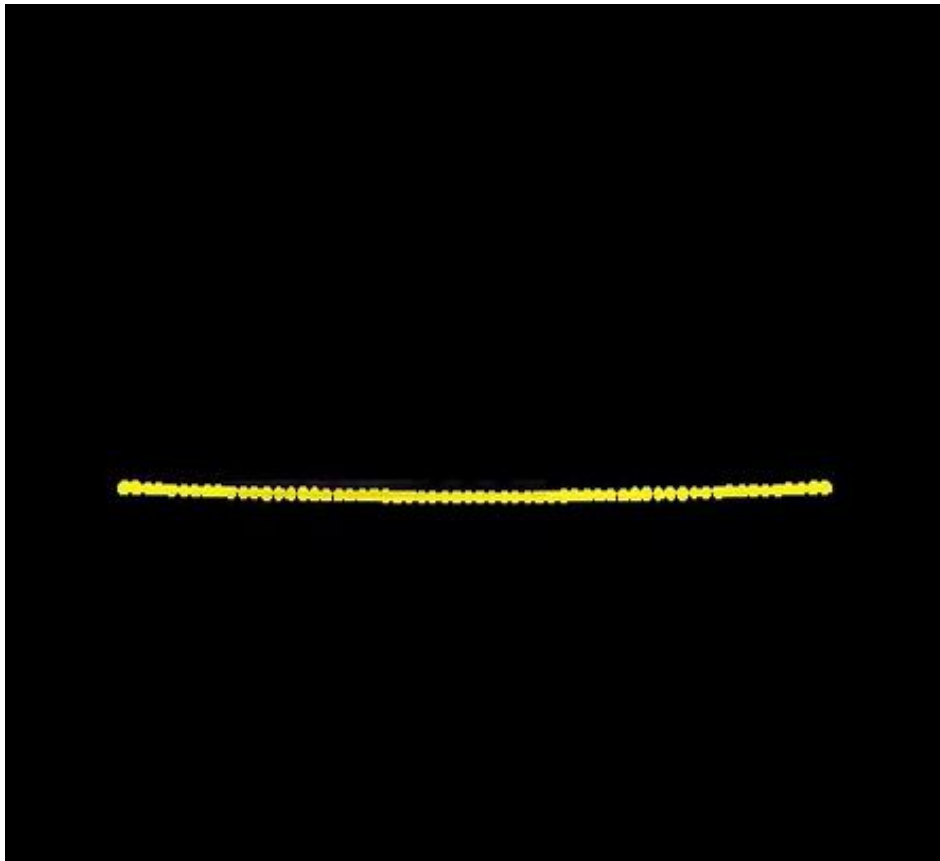


* SS, "Self-induced curvature in an internally loading peridynamic fiber," technical report SAND2022-5539

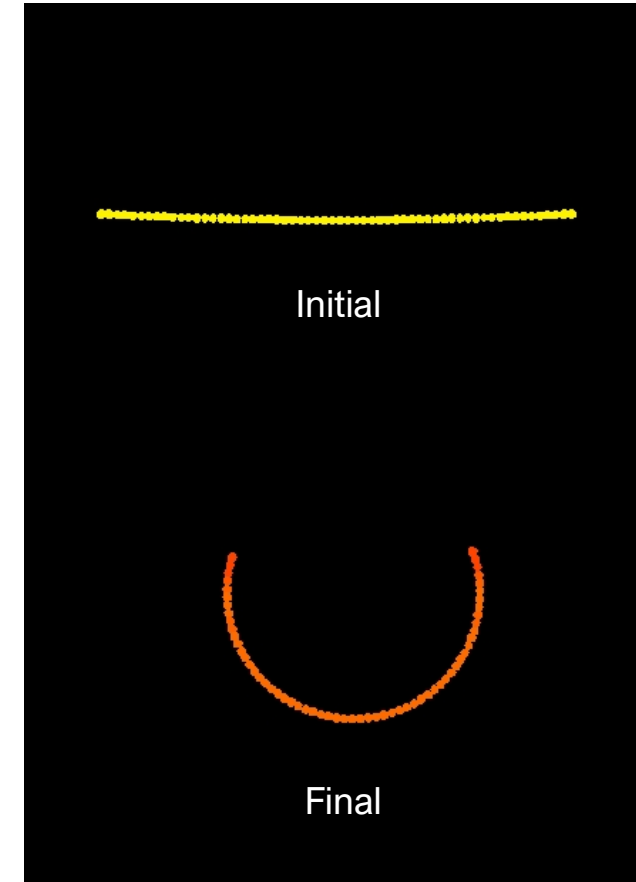
Emu simulation of an internally loaded fiber

- Internal loading is turned on suddenly.

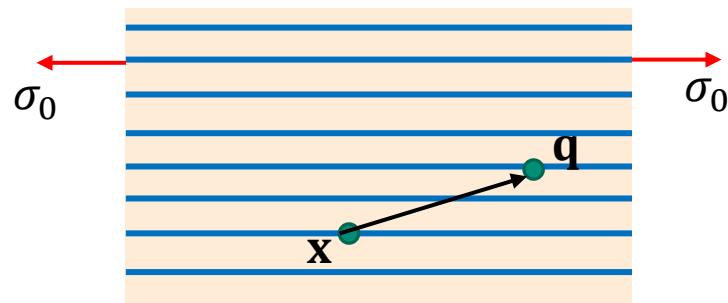
VIDEO



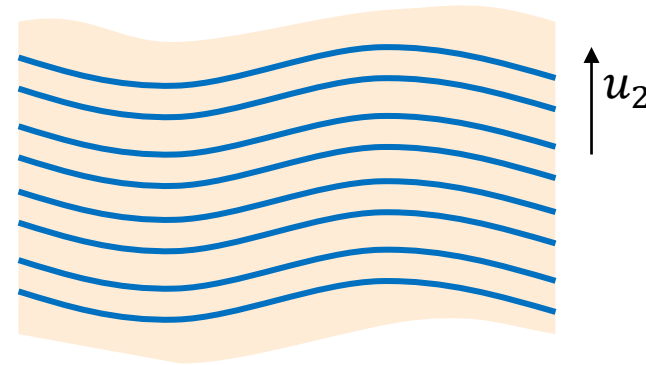
Colors show strain
(red = 0.01)



Homogenized model of many fibers: Kink bands in a fiber-reinforced composite



Fibers in a matrix
Remote loading in fibers is σ_0
Bonds can be either stiff or soft



Transverse wave

- The dispersion relation for transverse waves turns out to be

$$\rho\omega^2 = (\sigma_0 + \mu(k))k^2$$

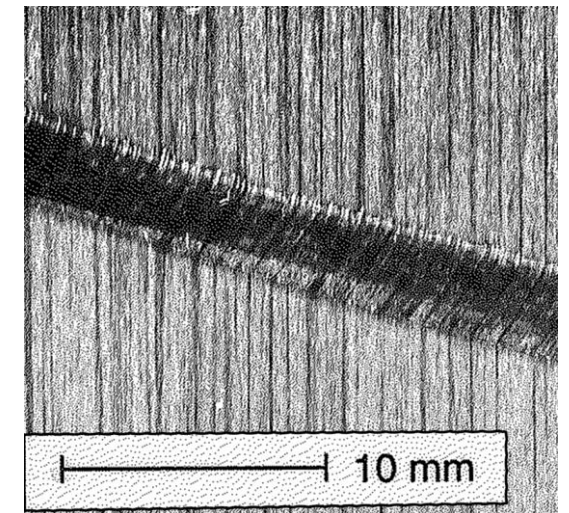
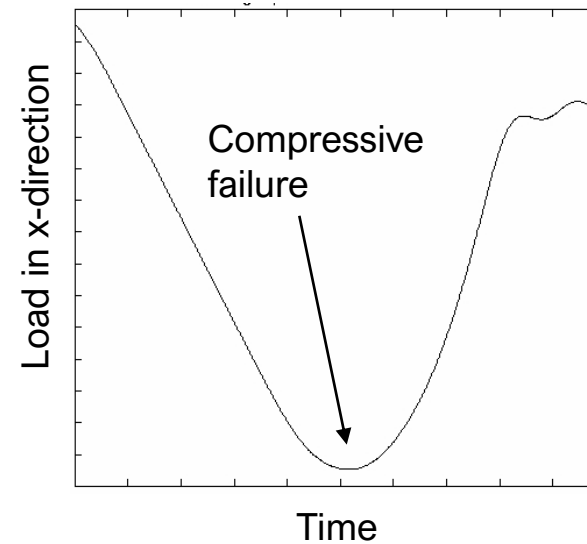
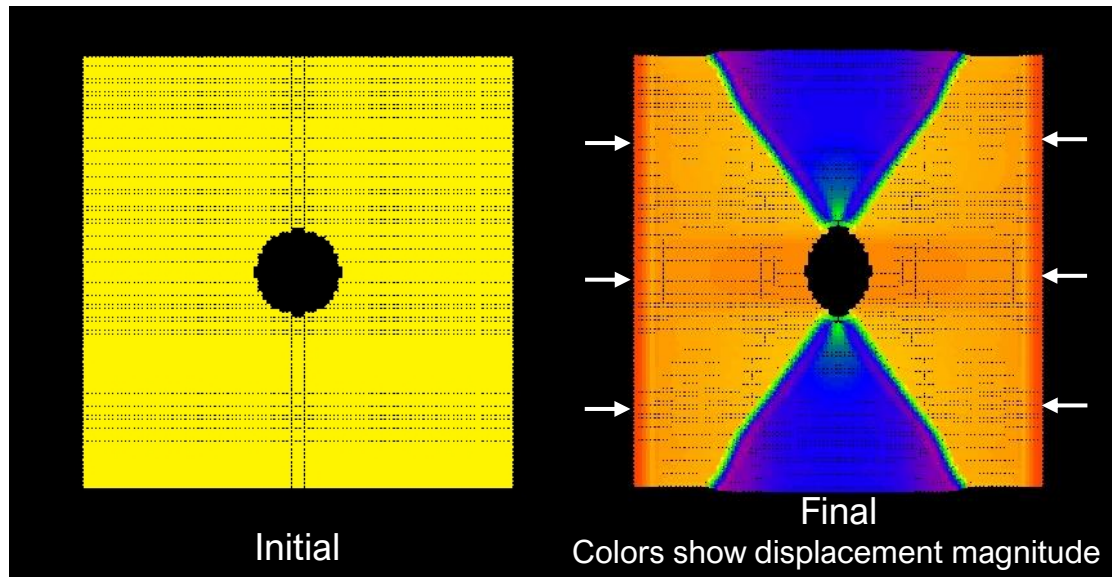
where $\mu(k)$ characterizes the matrix shear response.

- If the remote loading is compressive ($\sigma_0 < 0$) then ω can be imaginary.

Emu simulation of instability in compression in a composite



- Bonds in the horizontal direction are more stiff than the others.
- Anisotropic, microelastic material model.
- There is no damage (bond breakage) in the peridynamic model.

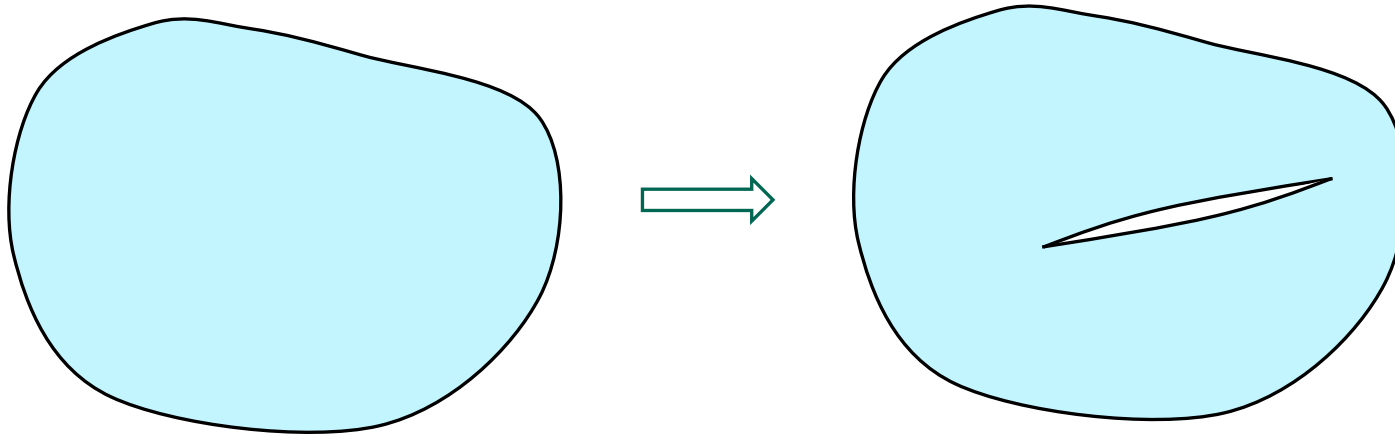


Kink band in a composite
Image: S.P.H. Skovsgard, thesis, 2019

Crack nucleation as a material instability



- A body is initially continuous.
- At some later time there is a discontinuity.
- What mathematical conditions need to exist for this to happen?
 - This question is not addressed by fracture mechanics, which assumes a pre-existing defect.

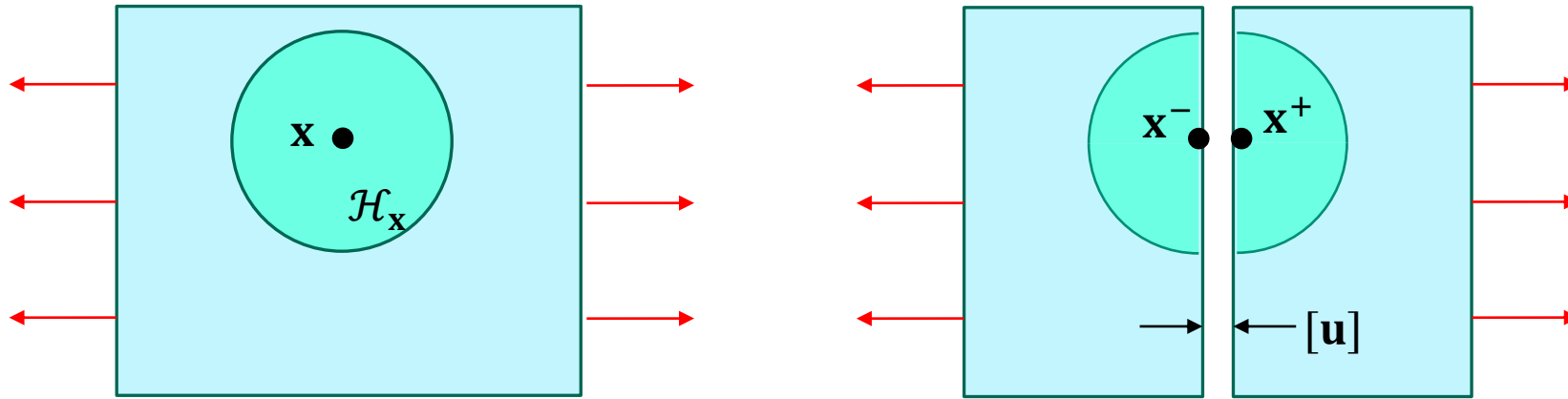


- SS, O. Weckner, E. Askari & F. Bobaru, *Int. J. Fracture* (2010)
- R.P. Lipton, R.B. Lehoucq & P.K. Jha, *J. Peridynamics & Nonlocal Modeling* (2019)

Perturbation by a jump



- Suppose a small virtual discontinuity is inserted into a body under load.
- Does the discontinuity grow or close up?
- Analyze the evolution of the jump $[u]$.



Perturbation by a jump



- From the momentum balance, find that

$$\rho[\ddot{\mathbf{u}}] = - \left(\int_{\mathcal{H}_x} \mathbf{C}(\xi) \, d\xi \right) [\mathbf{u}] = -\mathbf{P}[\mathbf{u}].$$

- The gap closes up if

$$[\ddot{\mathbf{u}}] \cdot [\mathbf{u}] < 0.$$

- If

$$\mathbf{u}_0 \cdot (\mathbf{P}\mathbf{u}_0) > 0 \quad \forall \mathbf{u}_0 \neq \mathbf{0}$$

then a crack cannot form.

- Let p_1, p_2, p_3 be the eigenvalues of \mathbf{P} (which is symmetric).
- *Stability index:*

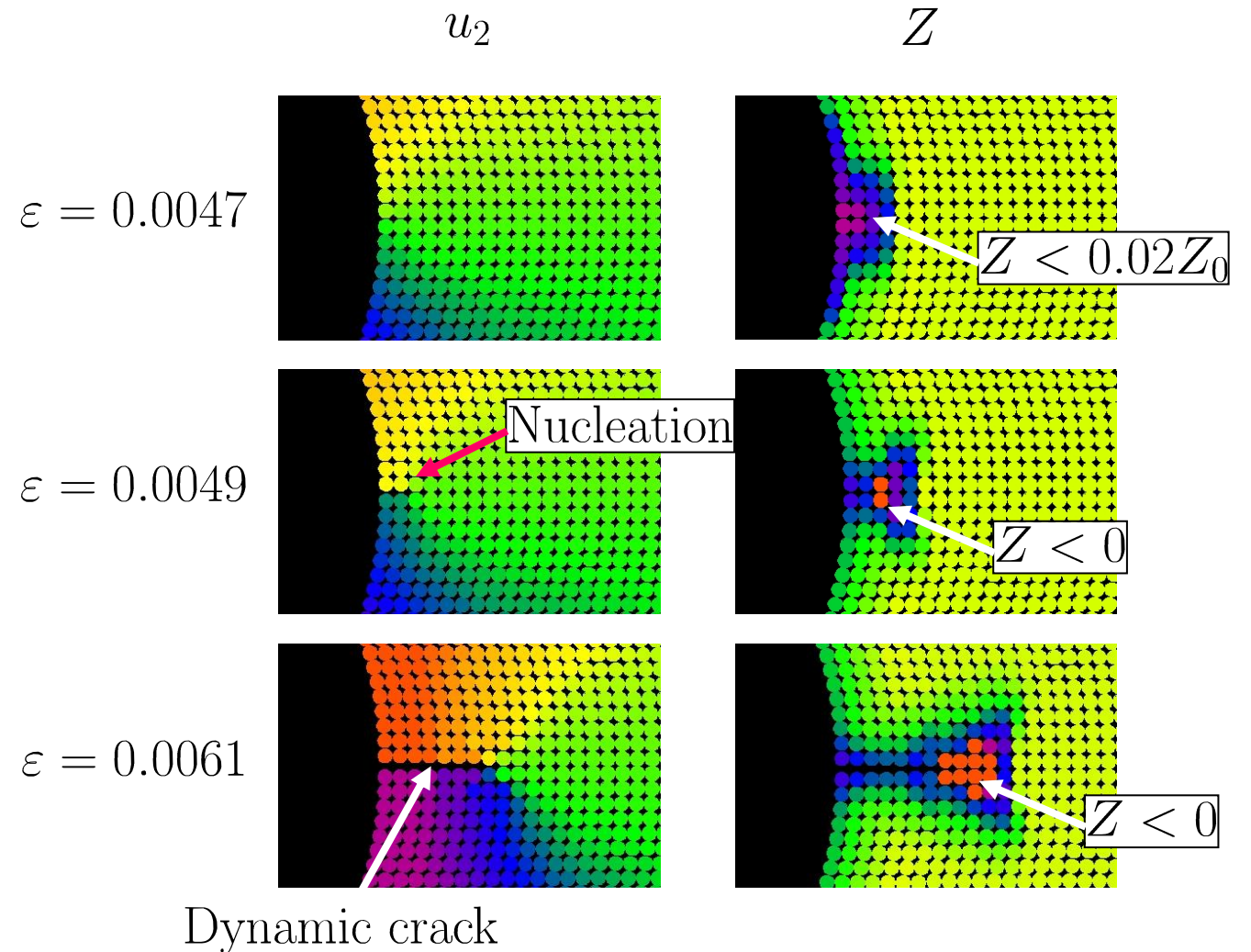
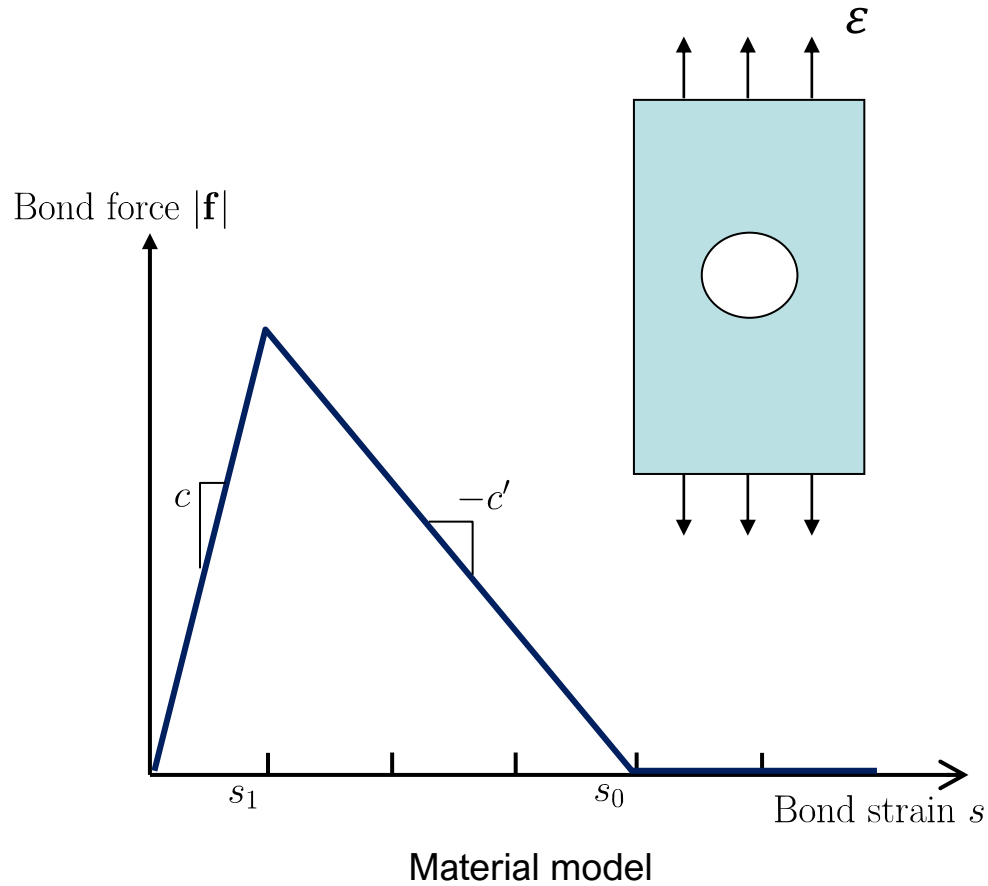
$$Z = \min\{p_1, p_2, p_3\}.$$

- If $Z \leq 0$, a crack can nucleate. The eigenvector gives the gap displacement.
- The orientation of the crack comes from the underlying stress field.

$Z < 0$ at the crack nucleation site and near a growing crack tip



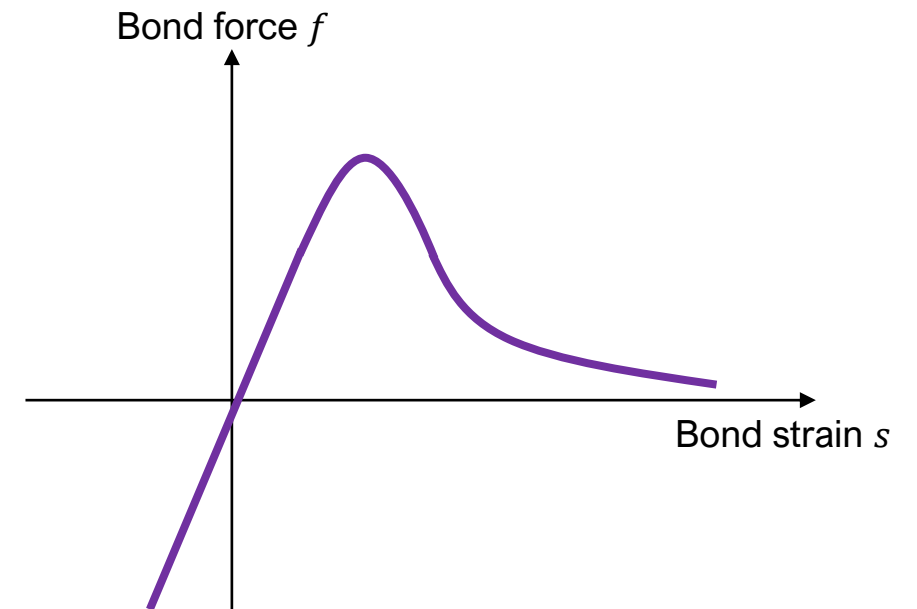
- There is a process zone near a growing crack tip within which $Z < 0$.



A nonlocal model of fracture can be well-posed (!)



- Nonconvex elastic peridynamic material.
- Points entering the unstable branch lead to creation of a jump in displacement.
- This discontinuity can grow through the body.
- As it grows, it consumes a definite amount of potential energy per unit crack area (Griffith crack).
- The whole problem can be well-posed.
- These results hold for both static and dynamic cases.

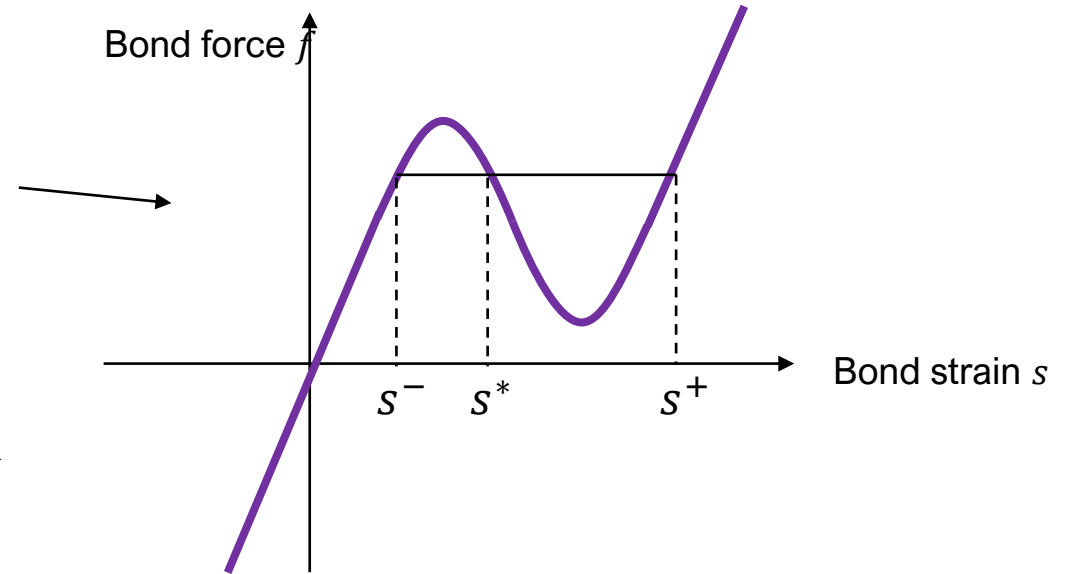
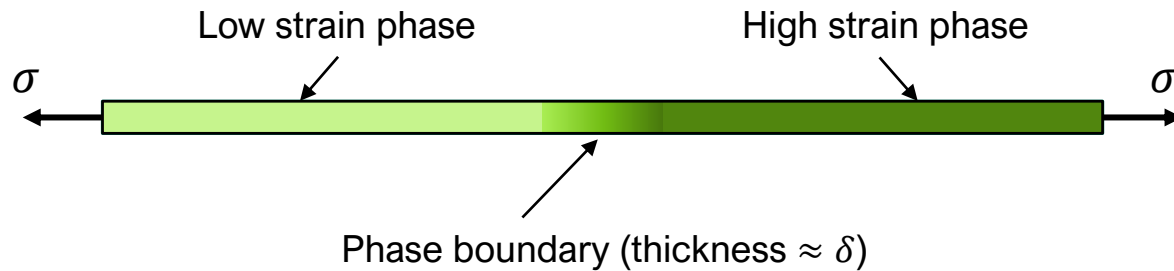


- R.P. Lipton, J. Elasticity (2014)
- R.P. Lipton, J. Elasticity (2016)
- P.K. Jha & R.P. Lipton, SIAM J. Numerical Analysis (2018)
- R. Lipton, E. Said & P.K. Jha - *Handbook of nonlocal continuum mechanics for materials and structures* (2018)
- R.P. Lipton, R.B. Lehoucq & P.K. Jha, *J. Peridynamics & Nonlocal Modeling* (2019)
- P.K. Jha & R.P. Lipton, Computer Methods in Applied Mechanics & Engineering (2019)
- R.P. Lipton & P.K. Jha, Nonlinear Differential Equations and Applications (2021)

Mechanically induced phase transformation



- Here's another type of nonconvex microelastic material:



- There are multiple bond strains for the same bond force.
- This creates the possibility of multiple phases in a bar in equilibrium.
- Within the phase boundary, there are always some bonds in the unstable part of the material model.

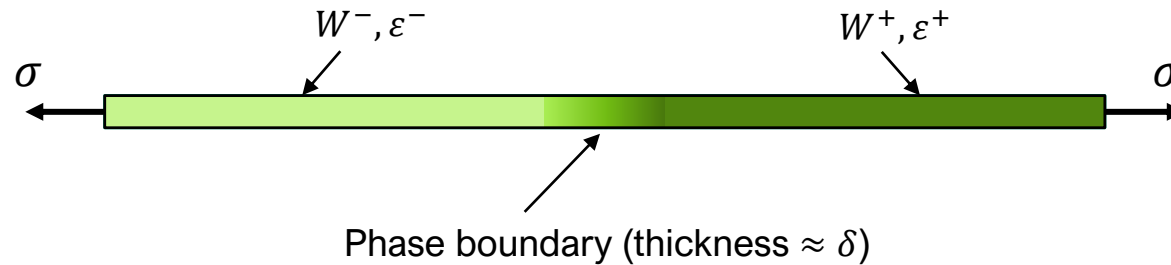
What conditions permit coexistent phases?

- In equilibrium, energy conservation implies

$$W^+ - W^- = (\varepsilon^+ - \varepsilon^-)\sigma$$

which is formally the same as in the local theory.

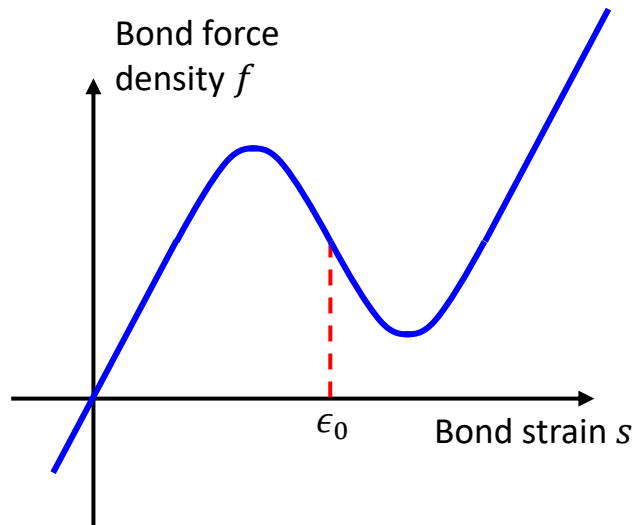
- Compare Weierstrass corner condition, Maxwell condition in the calculus of variations.



In peridynamics, a phase boundary contains potential energy

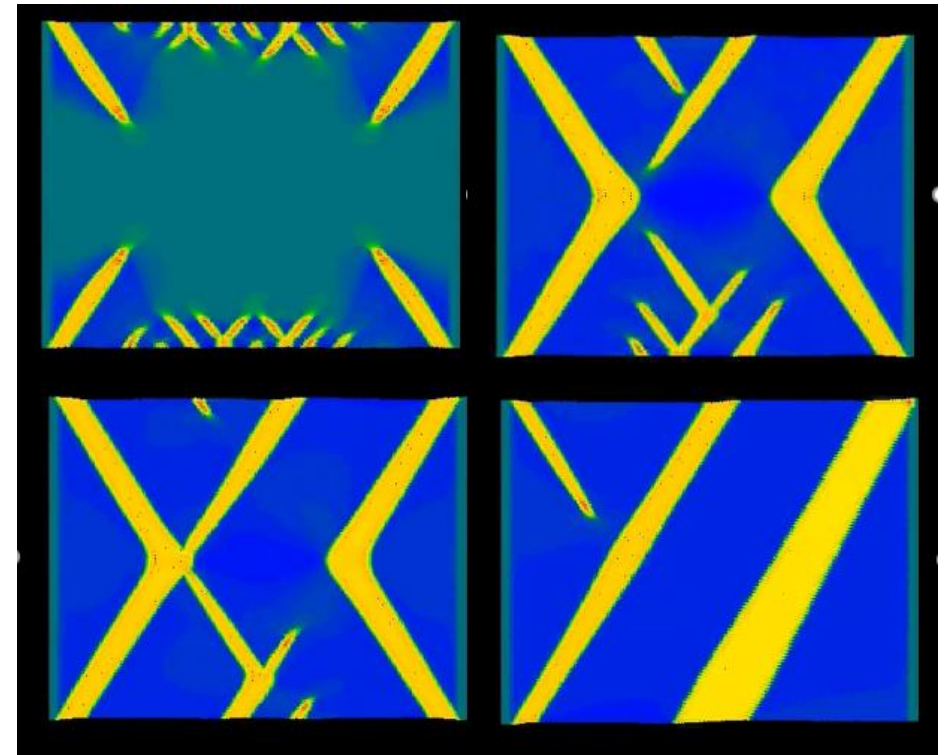


- The system evolves over time to reduce the total surface area of phase boundaries.
- We can use this to simulate microstructure evolution.



Microstructure evolution in a plate with initial strain ϵ_0

Colors show bond strain



Microstructure evolution over time



VIDEO

Microstructure evolution in a plate with initial strain ε_0
Colors show bond strain





- Nonlocality changes everything about material stability.
- We can do meaningful continuum mechanics within an unstable material.
- We can use this to simulate real phenomena:
 - Rate effect on bulk material strength
 - Fracture nucleation, Griffith criterion
 - Compressive failure modes in composites
 - Microstructure evolution