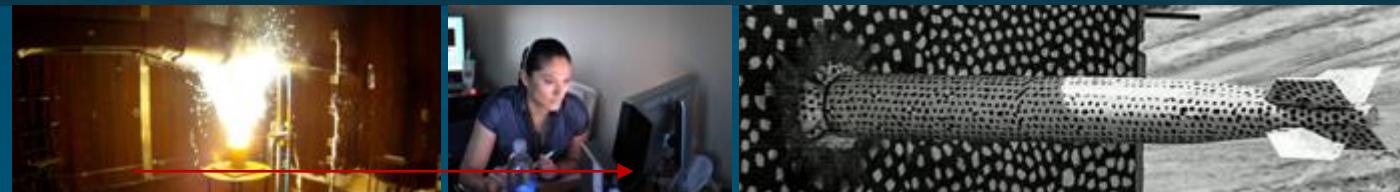




Sandia  
National  
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# Some aspects of material stability in peridynamics



Stewart Silling

June 21, 2022  
USNCTAM 2022  
Austin, TX

SAND2021-0000



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- Working virtually created many challenges.



Dear Boss,  
Can't do any work today. Cat on computer.



- But returning to the office is also creating challenges.



Dear Boss,  
Can't come to work today. Cat on pants.

# Outline



- **The nonlocal length scale affects all aspects of material stability.**
  - Failure kinetics: “Imaginary wave speeds” are useful.
  - Material and structural instability are related, leading to models for:
    - Kink bands in composites.
    - Self-shaping of fibers.
  - Fracture nucleation is a type of material instability.
  - Phase boundaries contain an unstable core.

# Some concepts of stability (local theory)



- Real wave speeds (Hadamard stability, strong ellipticity)
- Minimum potential energy
- Bifurcations
- Well-posedness
- Discontinuity in gradient (Ordinary ellipticity)

# Material vs. structural stability (local theory)



- Material instability:
  - Happens at a material point, triggered by local conditions
  - Not directly related to the geometry of the body
  - Example: Adiabatic shear band
- Structural instability:
  - Happens to the entire body collectively
  - Example: Buckling of a beam



Adiabatic shear band in aluminum  
image: Baxevanis et al.,  
[www.ima.umn.edu/materials/2008-2009/SP7.13-31.09/8186/ima.pdf](http://www.ima.umn.edu/materials/2008-2009/SP7.13-31.09/8186/ima.pdf)



Buckling of a column  
image: Klimchik  
[www.researchgate.net/figure/Examples-of-buckling-in-column-www-civildb-www-highline\\_fig11\\_281183936](http://www.researchgate.net/figure/Examples-of-buckling-in-column-www-civildb-www-highline_fig11_281183936)

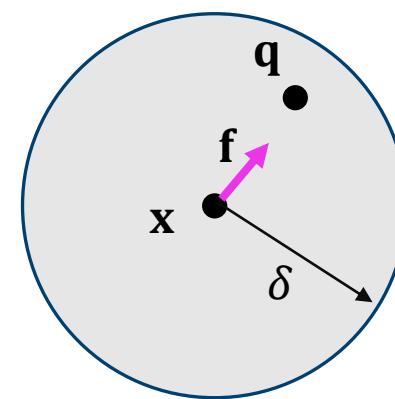
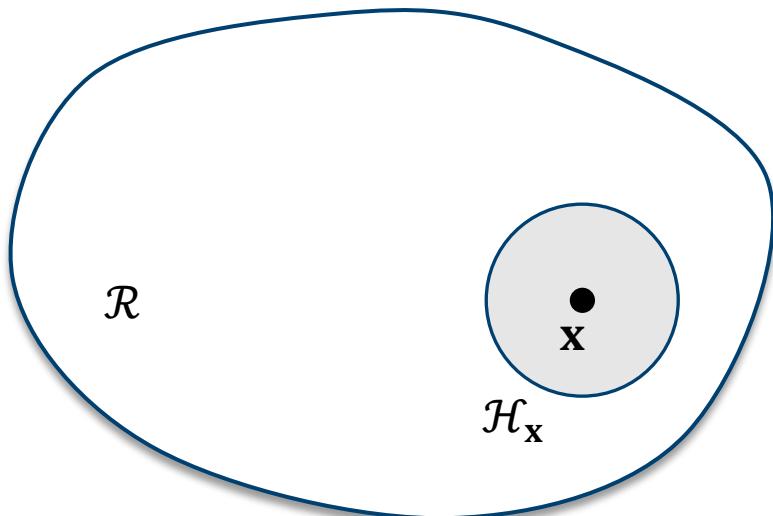
# Peridynamics background



- Peridynamic momentum balance in 3D:

$$\rho(\mathbf{x})\ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}_x} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) d\mathbf{q} + \mathbf{b}(\mathbf{x}, t) \quad \forall \mathbf{x} \in \mathcal{R}, t \geq 0.$$

- $\mathbf{f}$  is the *pairwise bond force density* of the *bond* from  $\mathbf{q}$  to  $\mathbf{x}$ .
- $\mathcal{H}_x$  is the *family* of  $\mathbf{x}$ , which is a ball centered at  $\mathbf{x}$  with radius  $\delta$  (the *horizon*).



## 8 Bond-based materials



- Later we will consider 3D deformations and bending of 1D-like structures.
- But for now, set

$$\mathbf{u} = u\mathbf{e}_1, \quad \mathbf{M} = \mathbf{e}_1, \quad \mathbf{f} = C(|q - x|)s\mathbf{e}_1$$

where  $C$  is a scalar function of the bond length called the *micromodulus* or *kernel*.

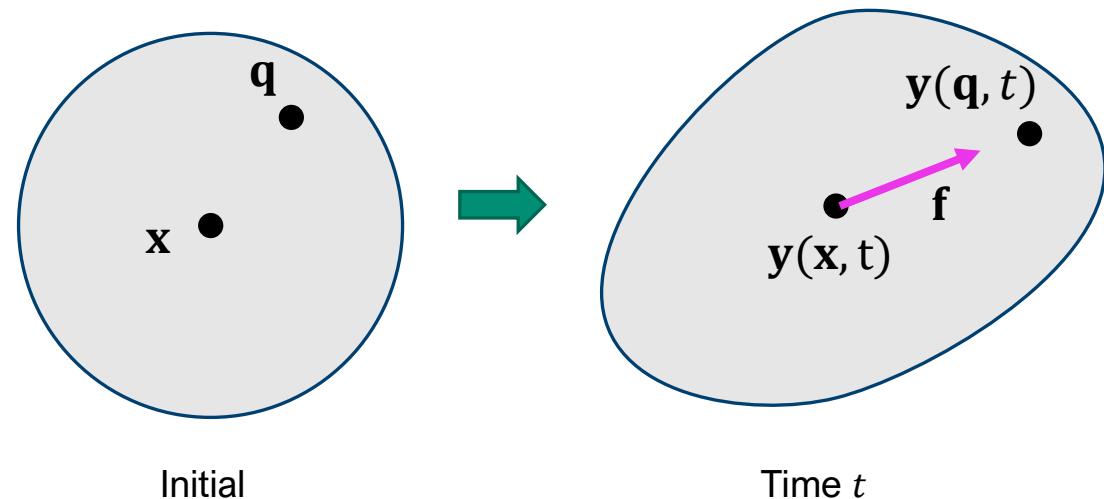
- The momentum balance is now

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi)(u(x + \xi) - u(x)) \, d\xi$$

(similar to Kunin's theory (1983)).

- $C$  must satisfy

$$C(-\xi) = C(\xi) \quad \forall \xi.$$



## 9 An intuitive notion of stability... and a mysterious tensor



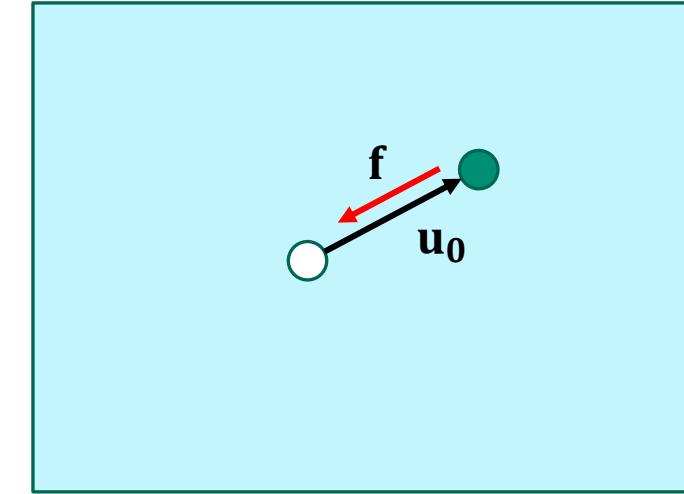
- Equilibrium:

$$\mathbf{L}(\mathbf{x}) + \mathbf{b}(\mathbf{x}) = \mathbf{0}, \quad \mathbf{L}(\mathbf{x}) = \int_{\mathcal{H}_x} \mathbf{C}(\xi)(\mathbf{u}(\mathbf{x} + \xi) - \mathbf{u}(\mathbf{x})) \, d\xi$$

where  $\mathbf{L}$  is the force density on  $\mathbf{x}$ ,

- Suppose we “carve out” a small volume surrounding a point  $\mathbf{x}$  and displace it by  $\mathbf{u}_0$ .
- The net force density on the small volume is:

$$\mathbf{L} = -\mathbf{P}\mathbf{u}_0, \quad \mathbf{P} = \int_{\mathcal{H}_x} \mathbf{C}(\xi) \, d\xi.$$



where  $\mathbf{P}$  is the *single point response tensor*.

- *Single point stability*: If

$$\mathbf{u}_0 \cdot (\mathbf{P}\mathbf{u}_0) > 0 \quad \forall \mathbf{u}_0$$

then the particle always gets pushed back toward where it started. Otherwise  $\mathbf{x}$  can fly off to  $\infty$ .

Comparable statement in the local theory:  
 $c_{ijkl}\epsilon_{ij}\epsilon_{kl} > 0 \quad \forall \epsilon$

# Simplify further to 1D, linear microelastic

- Later we will consider 3D deformations and bending of 1D-like structures.
- But for now, set

$$\mathbf{u} = u\mathbf{e}_1, \quad \mathbf{M} = \mathbf{e}_1, \quad \mathbf{f} = C(|q - x|)s\mathbf{e}_1$$

where  $C$  is a scalar function of bond length called the *micromodulus* or *kernel*.

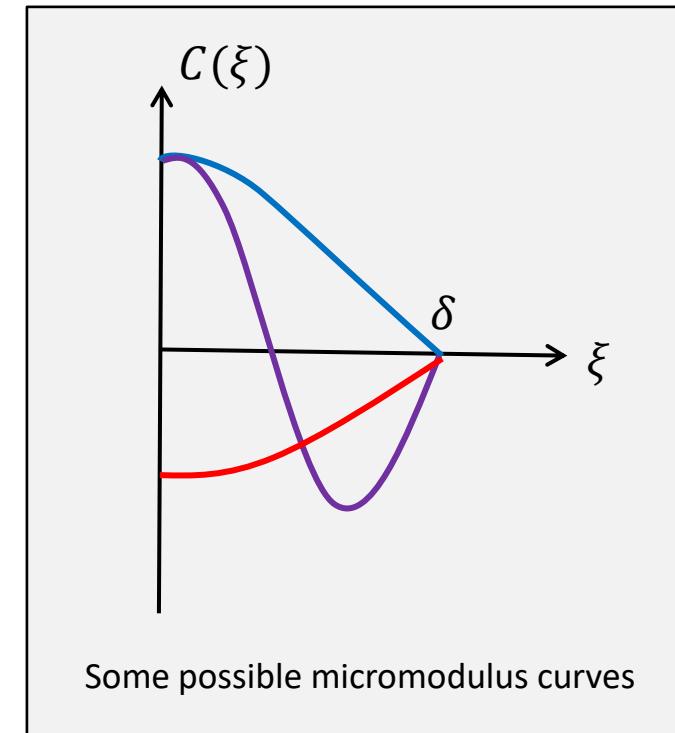
- The momentum balance is now

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi)(u(x + \xi) - u(x)) \, d\xi$$

(similar to Kunin's theory (1983)).

- $C$  must satisfy

$$C(-\xi) = C(\xi) \quad \forall \xi.$$



# Linear waves: Dispersion curves



- Assume a wave of the form

$$u(x, t) = A e^{i(kx - \omega t)}$$

where  $A$ =amplitude,  $k$ =wavenumber,  $\omega$ =frequency.

- Equation of motion:

$$\rho \ddot{u}(x, t) = \int_{-\delta}^{\delta} C(\xi) [u(x + \xi, t) - u(x, t)] d\xi$$

- leads to

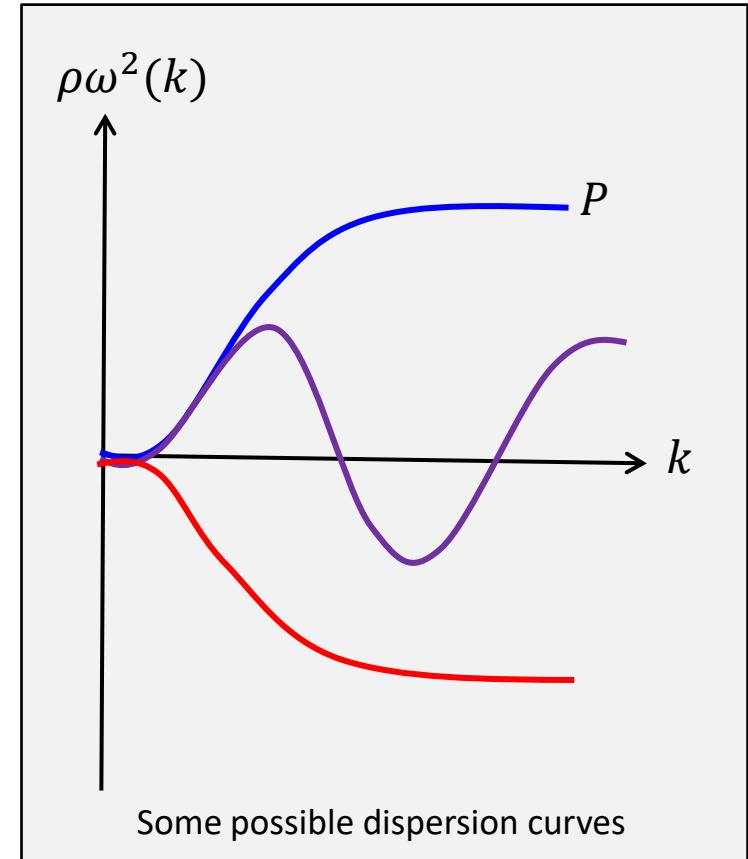
$$-\rho \omega^2 = \int_{-\delta}^{\delta} C(\xi) [e^{ik\xi} - 1] d\xi$$

- Therefore the dispersion relation is

$$\omega(k) = \sqrt{\frac{P - \bar{C}(k)}{\rho}}, \quad P = \bar{C}(0)$$

where  $\bar{C}(k)$  is the Fourier transform of  $C(\xi)$ .

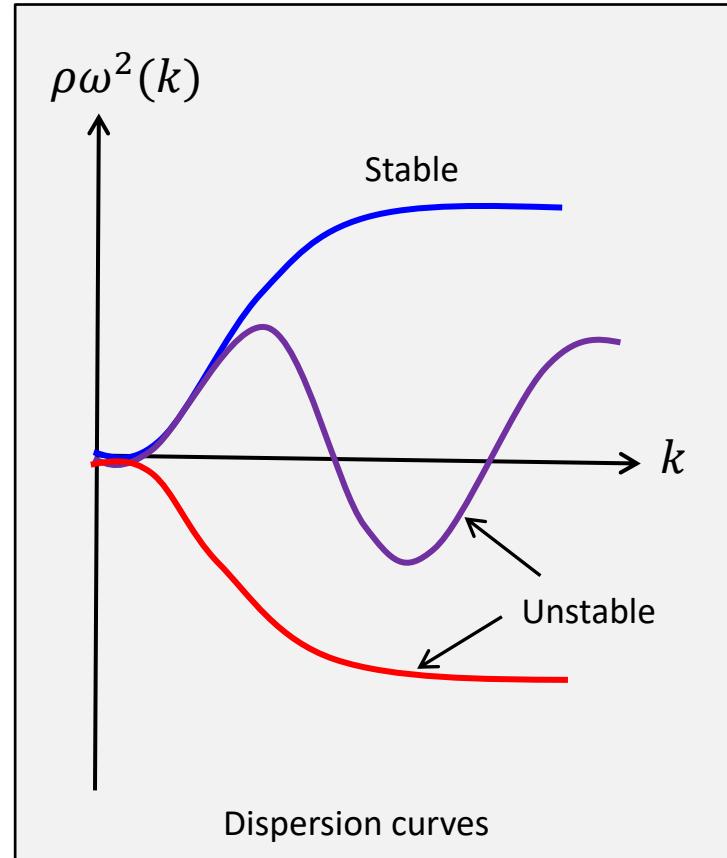
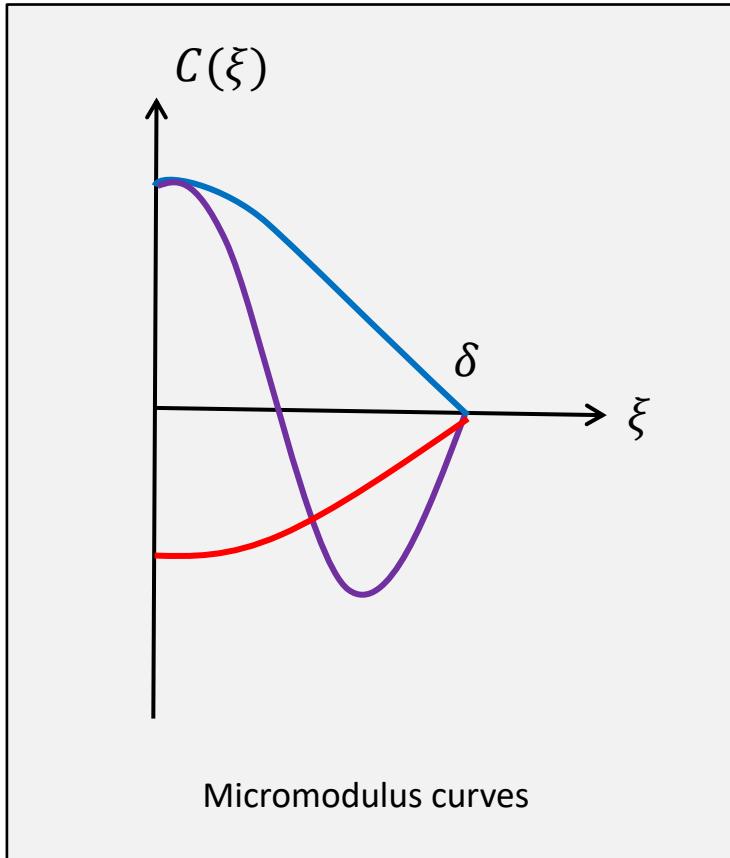
We've seen this before!



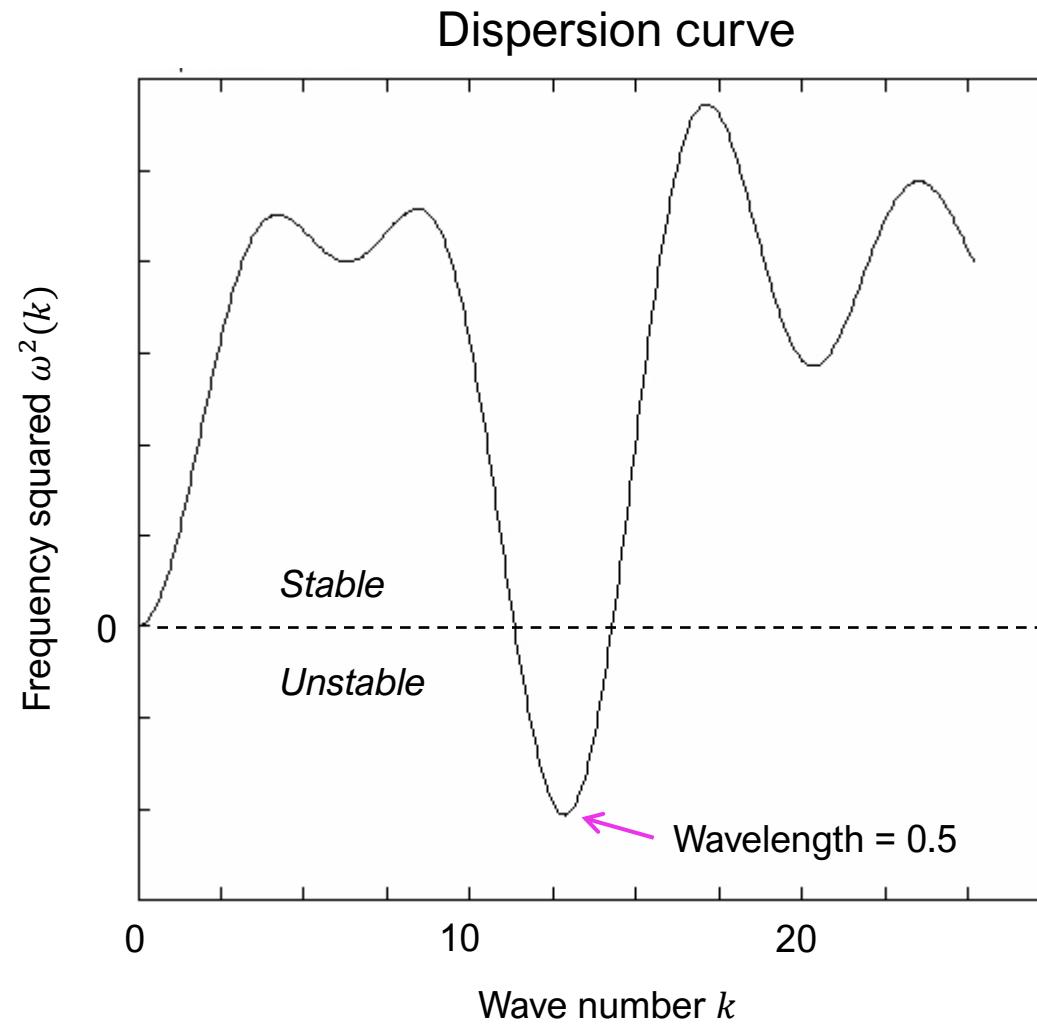
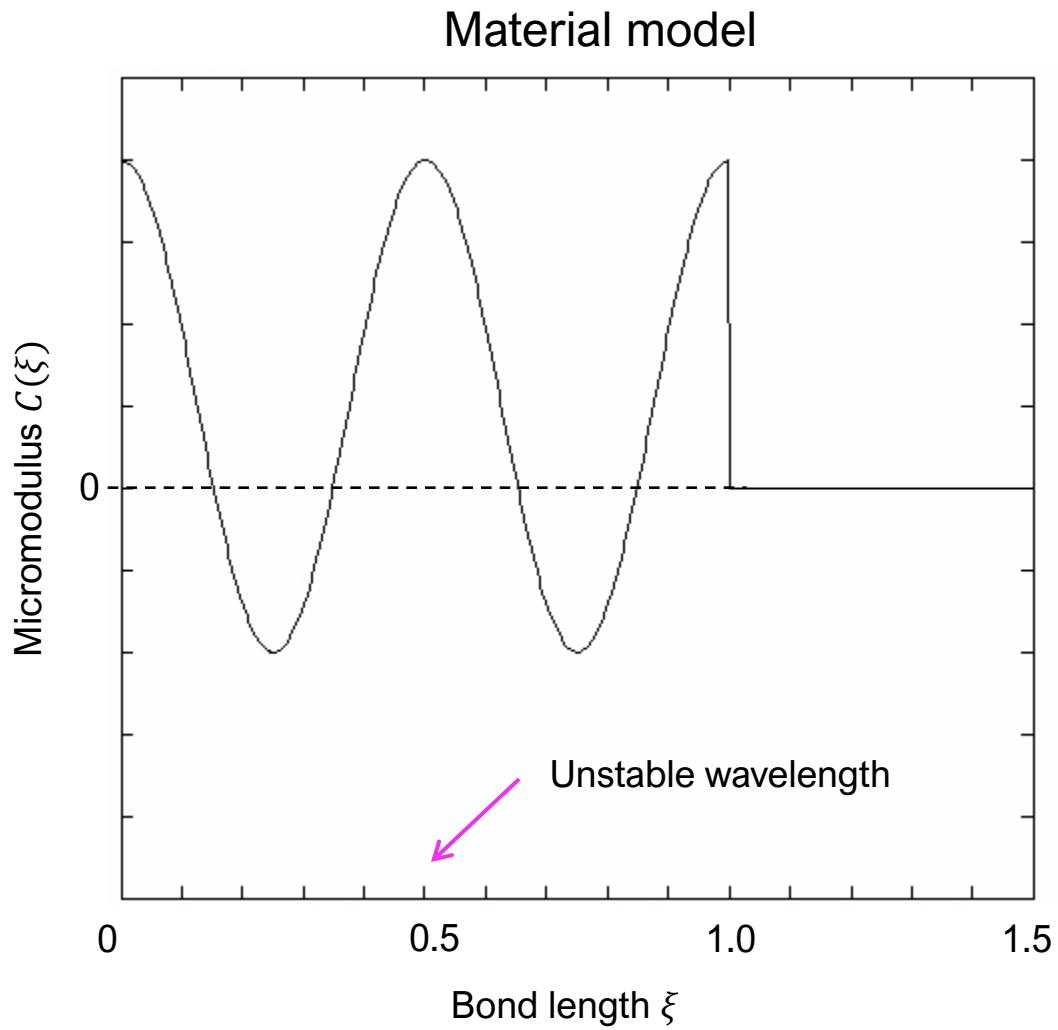
# Dispersion and material stability



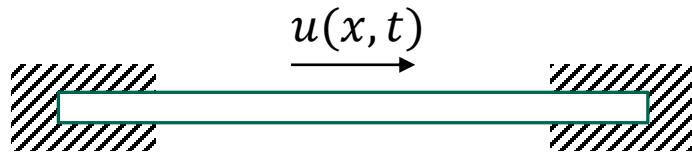
- Purple and red materials have “imaginary wave speeds”  $c = \omega/k$ .
  - Red: all wavenumbers are unstable.
  - Purple: only some wavenumbers are unstable.



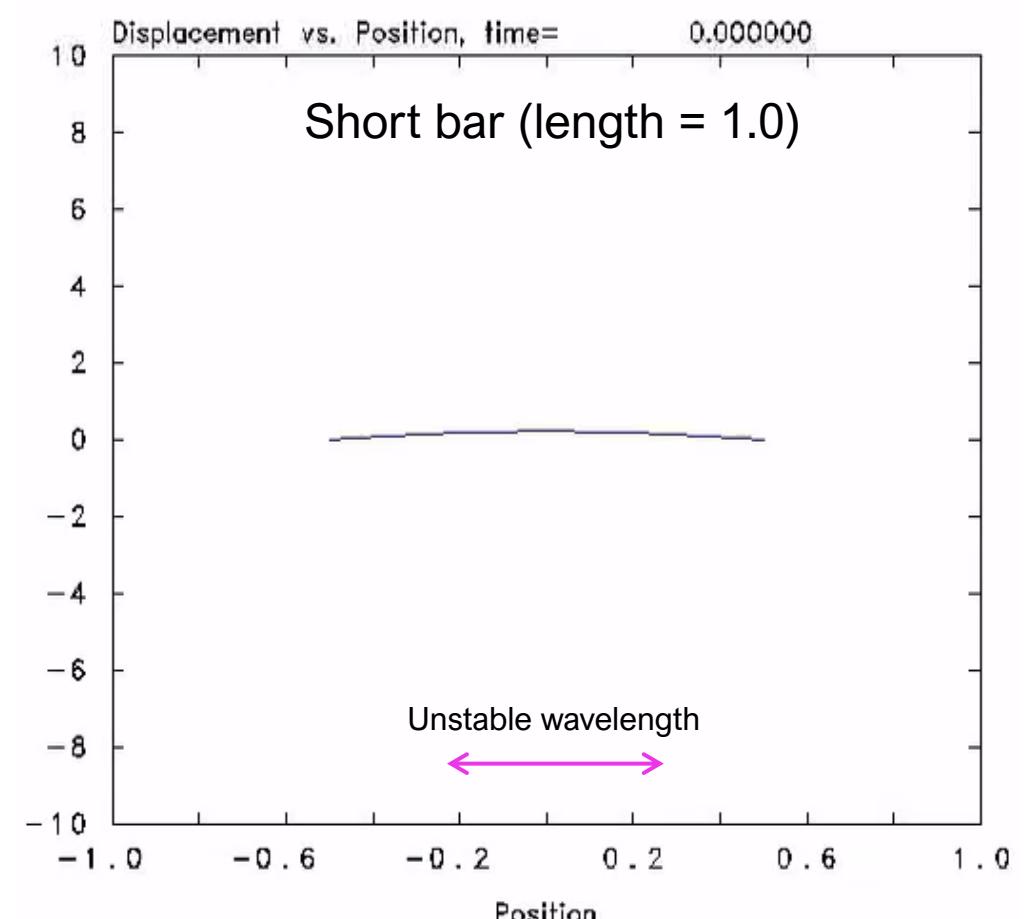
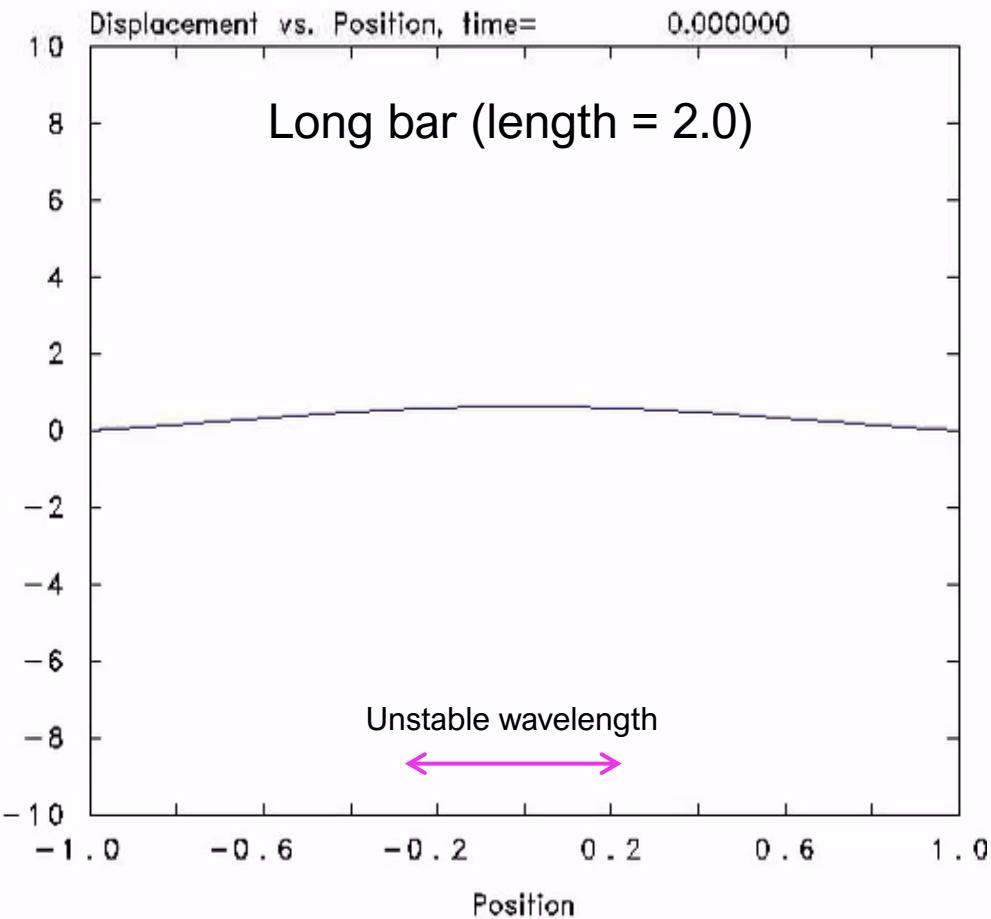
# Example: A material with a narrow band of unstable wavenumbers



# This material is stable in some geometries but unstable in others



VIDEOS



# Minimum potential energy is related to wave speed



- Total potential energy in a bounded body:

$$\Phi_{\mathbf{u}} = \frac{1}{2} \iint w(\mathbf{x}', \mathbf{x}) \, d\mathbf{x}' \, d\mathbf{x} + \int \mathbf{b} \cdot \mathbf{u} \, d\mathbf{x}$$

where  $w$  is the bond energy (*micropotential*).

- Consider a set of deformations parameterized by  $\varepsilon$ :

$$\mathbf{u} = \mathbf{u}_0 + \varepsilon \mathbf{v}$$

where  $\mathbf{v}$  is a vector field.

- Stationary  $\Phi$ :

$$\frac{d\Phi}{d\varepsilon} = 0 \quad \forall \mathbf{v}$$

leads to the equilibrium equation that is satisfied by  $\mathbf{u}_0$

$$\int \mathbf{f}(\mathbf{x}', \mathbf{x}) \, d\mathbf{x}' + \mathbf{b} = \mathbf{0} \quad \forall \mathbf{x}.$$

## Minimum potential energy is related to wave speed, ctd.



- Now require  $\Phi$  to be a minimum as well as stationary:

$$\frac{d^2\Phi}{d\varepsilon^2} > 0$$

for all  $\mathbf{v}$  except rigid motions.

- Leads to

$$\int \int (\mathbf{v}' - \mathbf{v}) \cdot \mathbf{C}(\xi) (\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} > 0 \quad \forall \mathbf{v}$$

where

$$\mathbf{C} = \frac{\partial \mathbf{f}}{\partial \boldsymbol{\eta}}, \quad \boldsymbol{\eta} := \mathbf{u}' - \mathbf{u}.$$

# Minimum energy implies real wave speeds



- Suppose  $\Phi$  is minimized by  $\mathbf{u}_0$ . Let  $\mathbf{v}$  be a standing wave (eigenmode, vibrational mode).

$$\int \mathbf{C}(\xi)(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' + \rho\omega^2 \mathbf{v} = \mathbf{0}.$$

- Multiply through by  $\mathbf{v}$  and integrate over  $\mathbf{x}$ :

$$\int \int \mathbf{v} \cdot \mathbf{C}(\xi)(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} + \rho\omega^2 \int \mathbf{v} \cdot \mathbf{v} \, d\mathbf{x} = 0.$$

- After some manipulations:

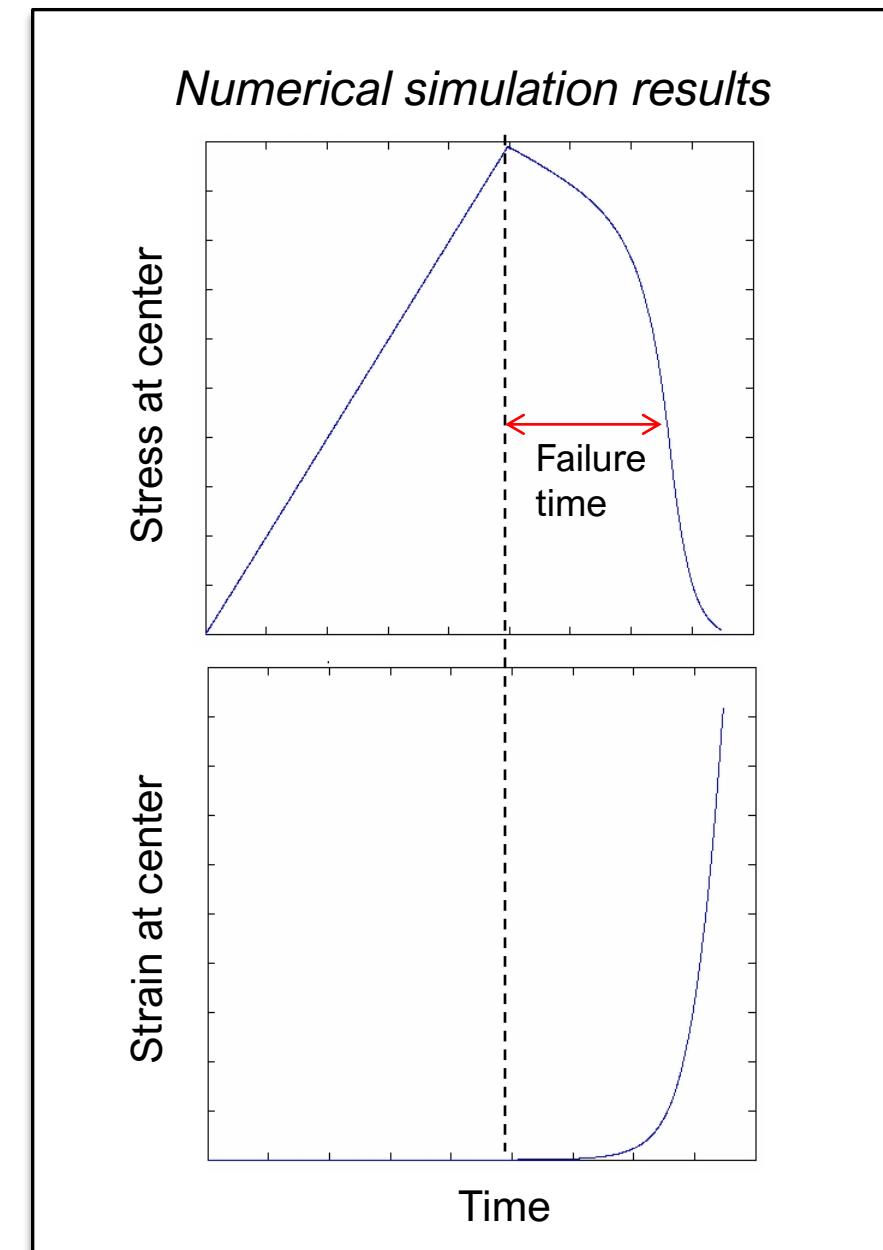
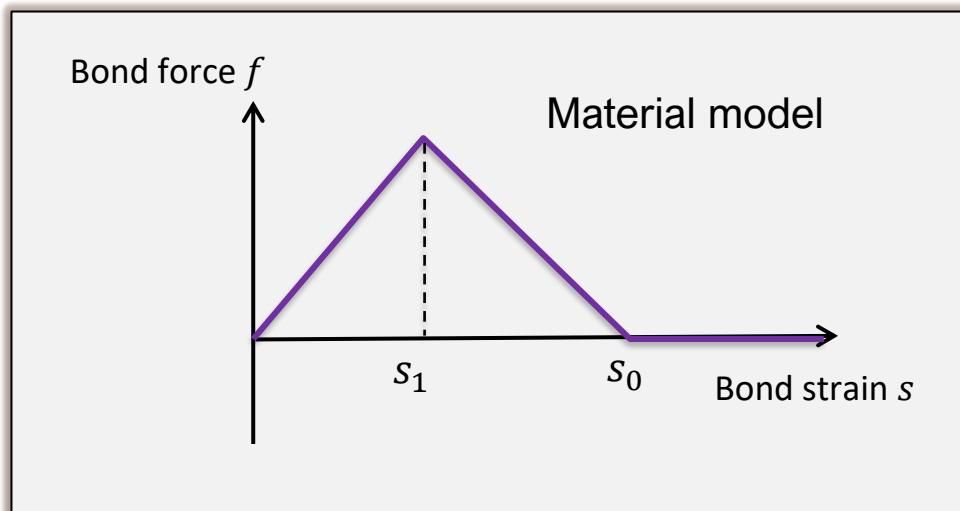
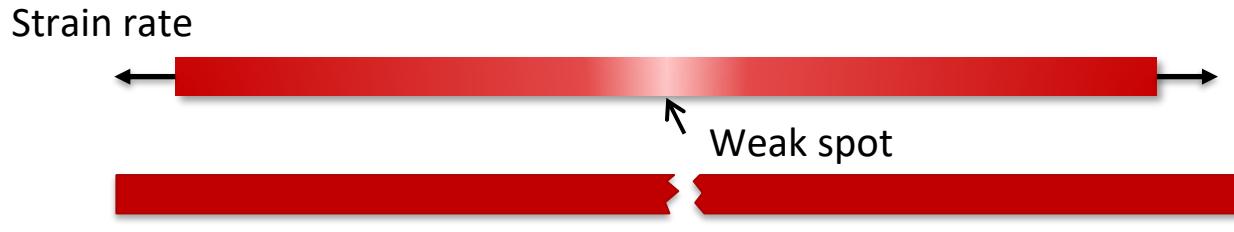
$$\int \int (\mathbf{v}' - \mathbf{v}) \cdot \mathbf{C}(\xi)(\mathbf{v}' - \mathbf{v}) \, d\mathbf{x}' \, d\mathbf{x} - 2\rho\omega^2 \int \mathbf{v} \cdot \mathbf{v} \, d\mathbf{x} = 0.$$

- But we already know that the  $\int \int$  is positive.
- Conclude  $\omega^2 > 0$ . So the wave speeds are real.

# Failure kinetics: How much time does it take for material to fail?



- Example: Stretching of a bar with an unstable material model.
  - Bar is stretched from the ends at a constant rate.
  - The bond force vs. strain curve has a descending branch.
  - What happens?



# Failure kinetics: Unstable waveforms grow exponentially but at a finite rate



- Initial data in the infinite bar:

$$u(x, 0) = A \cos kx, \quad \dot{u}(x, 0) = 0 \quad \forall x.$$

- If  $\omega^2(k) > 0$ :

$$u(x, t) = \frac{A}{2} [\cos(kx - \omega(k)t) + \cos(kx + \omega(k)t)].$$

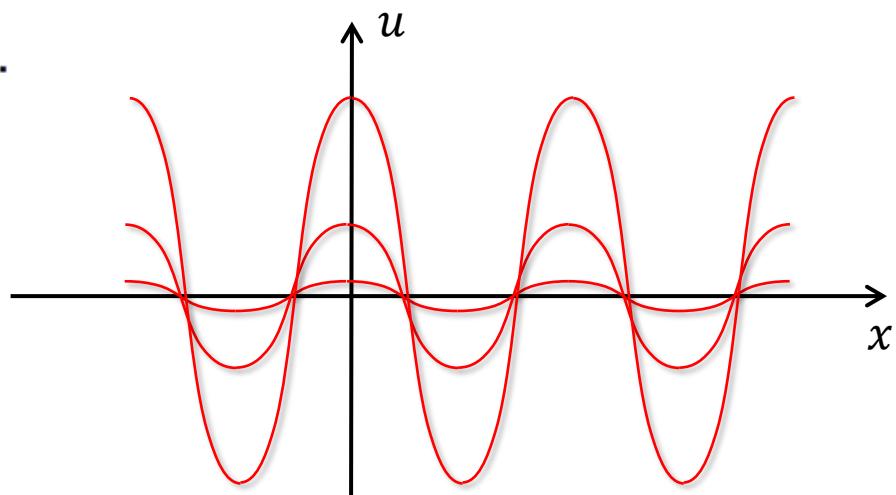
- If  $\omega^2(k) < 0$ :

$$u(x, t) = A \cos(kx) \cosh(\lambda(k)t)$$

where

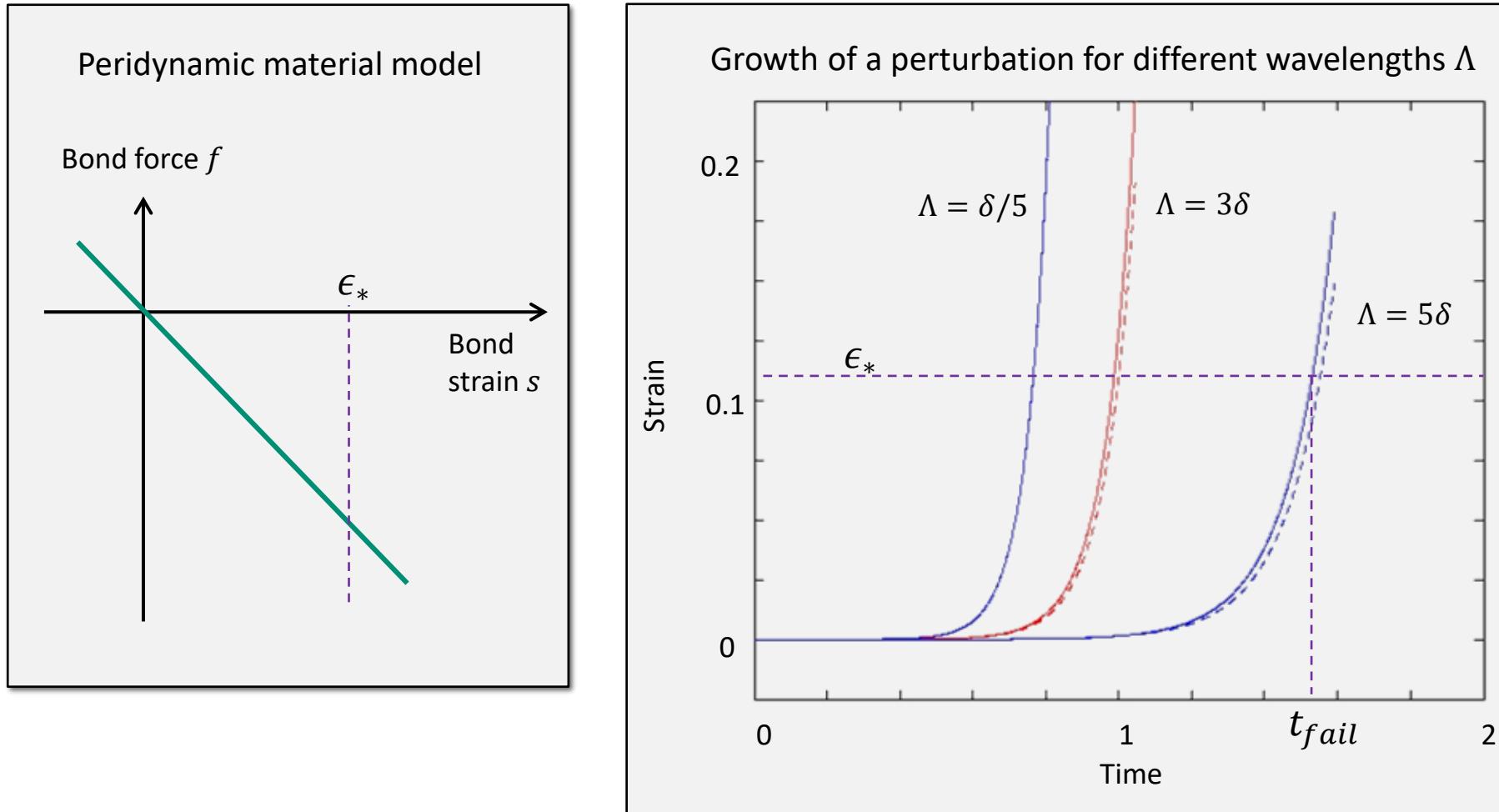
$$\lambda(k) = \sqrt{-\omega^2} \quad \text{real.}$$

- $\lambda$  is called the *blow-up rate*.



# How much time does it take for an unstable waveform to grow?

- Suppose the material “fails” when the local strain  $u_x$  exceeds some given value  $\epsilon_*$ .

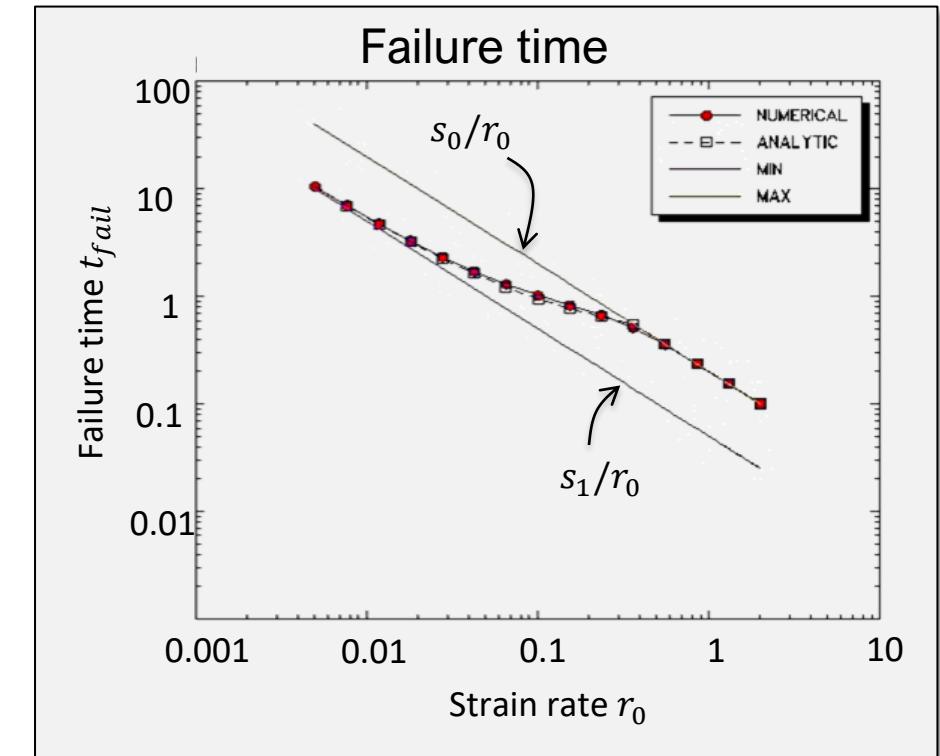
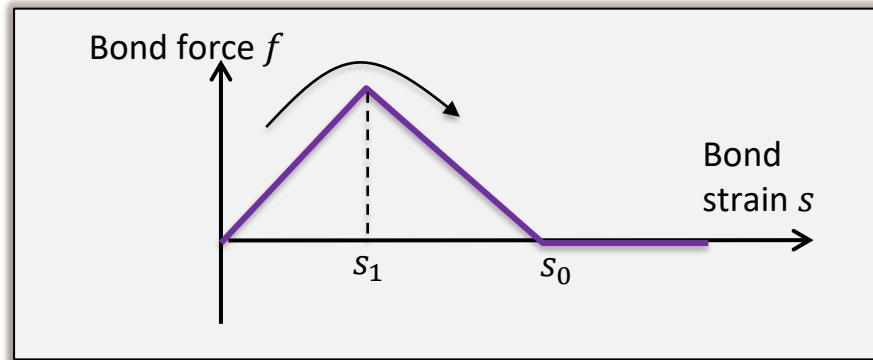
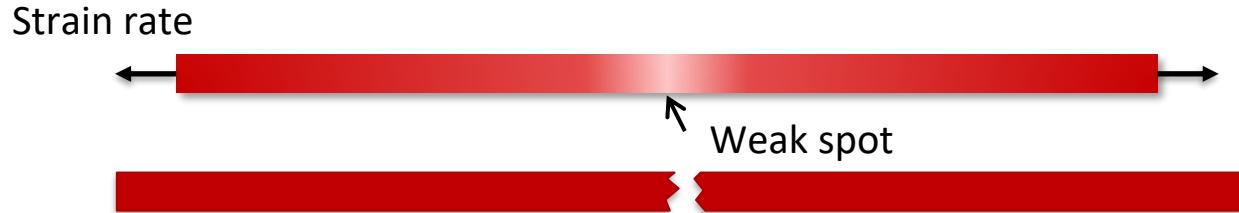


# Stretching of a bar: Compute the time to failure



$$t_{fail} \approx \min \left\{ \frac{s_0}{r_0}, \left[ \frac{s_1}{r_0} + \frac{1}{\lambda_\infty} \log \left( \frac{2(s_0 - s_1)}{h_0(s_1 + r_0/\lambda_\infty)} \right) \right] \right\}$$

$$\lambda_\infty = \sqrt{\frac{-P}{\rho}}$$



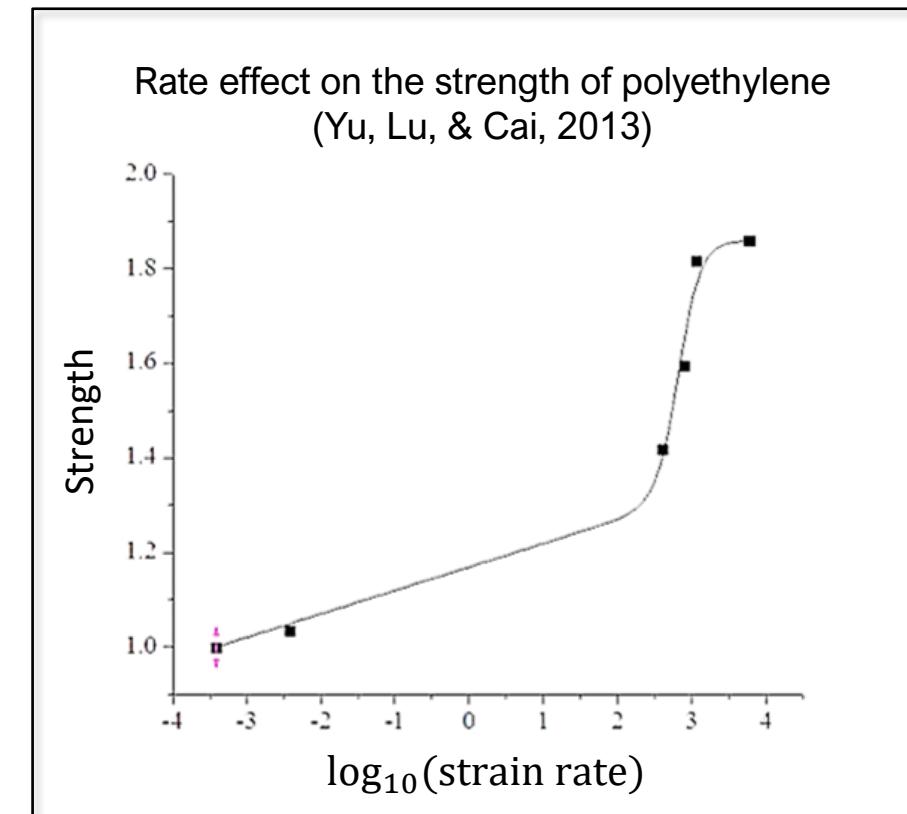
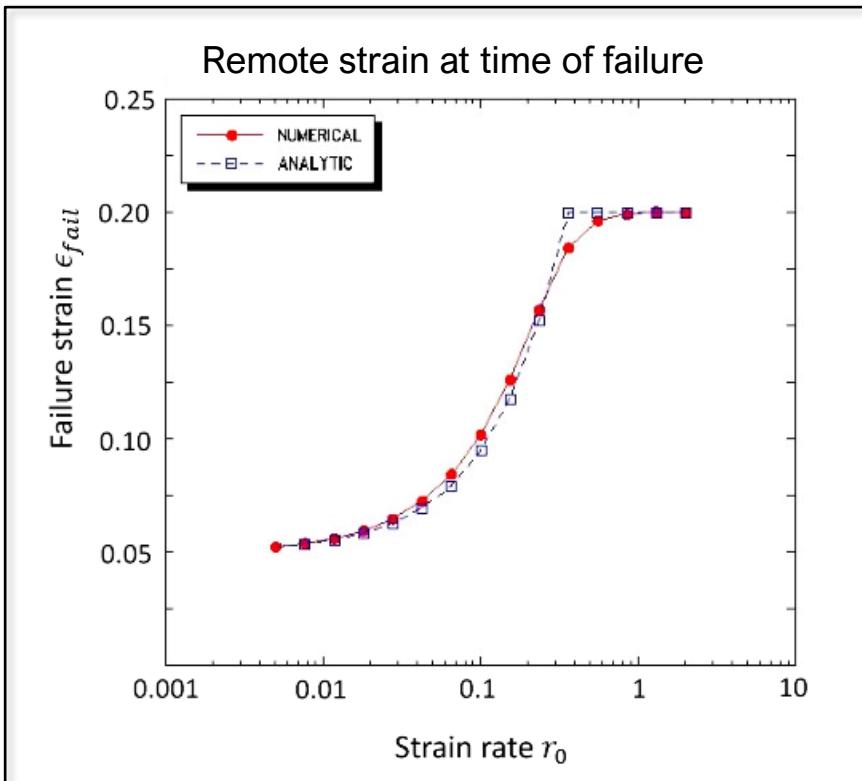
# Stretching of a bar: We arrive at a macroscopic rate effect



- It takes time for the bar to fail even after some bonds have crossed the peak.
- Meanwhile, the remote strain is still increasing.
- Result is that higher strain rates lead to higher macroscopic failure strain

$$\epsilon_{fail} = r_0 t_{fail}$$

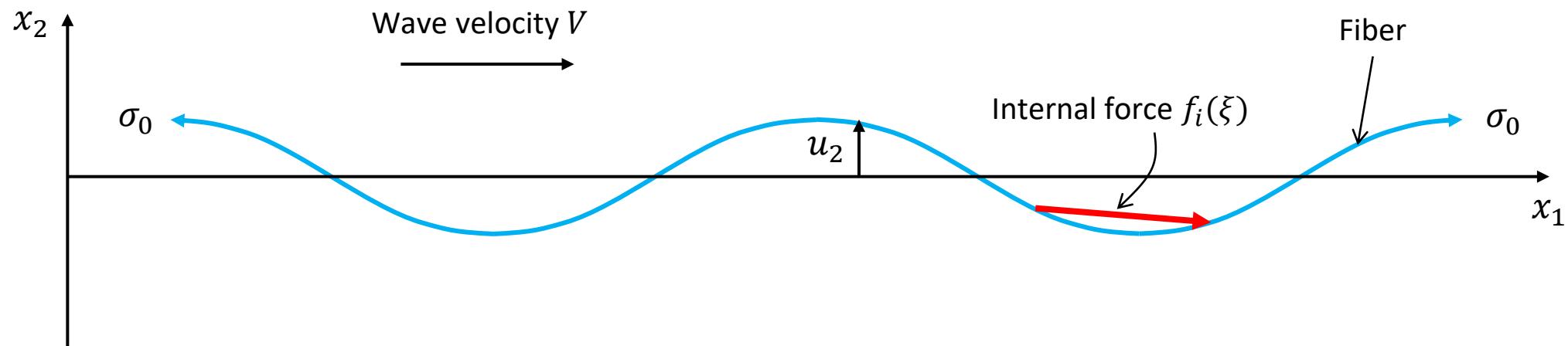
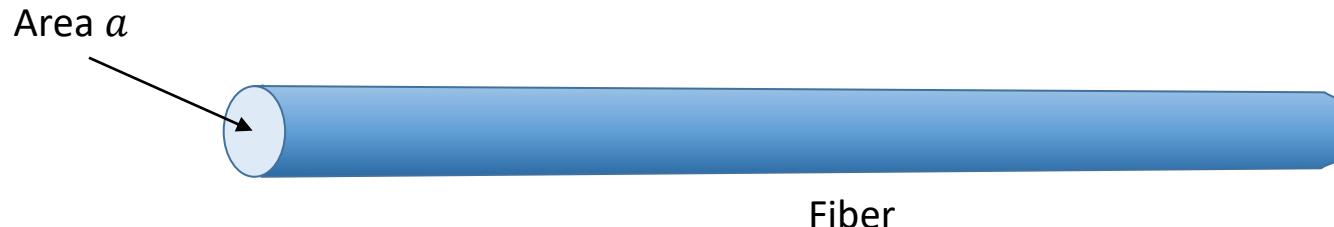
- Many real materials show a similar trend.



# Instability due to internal loading in a fiber



- String made of microelastic material
- Constant long-range forces between material points
- Allow rotations (unlike true 1D)
- Study transverse waves and their stability



# Internal loading in a fiber accounting for bond rotations



- Recall the 3D momentum balance:

$$\rho(\mathbf{x})\ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}_x} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) \, d\mathbf{q} + \mathbf{b}(\mathbf{x}, t) \quad \forall \mathbf{x} \in \mathcal{R}, \, t \geq 0.$$

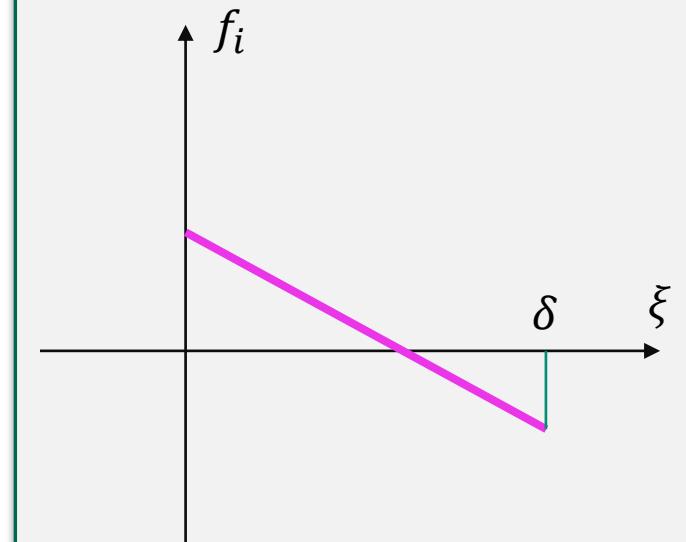
- Allow for bond rotation.

$$\mathbf{f}(\mathbf{q}, \mathbf{x}, t) = (C(\xi)s + f_i(\xi))\mathbf{M}$$

where  $s$  is the bond strain and  $f_i$  is the prescribed internal force density in the bond  $\xi$ .

- Assume that  $f_i$  is self-equilibrated (no net axial stress).

Internal loading as a function of bond length





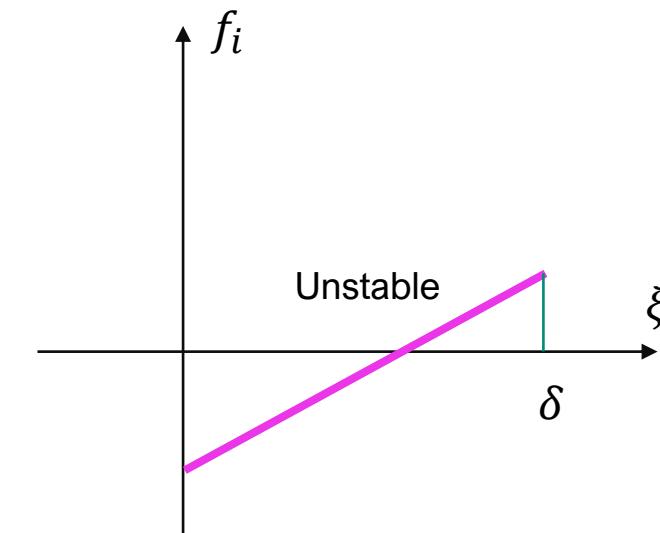
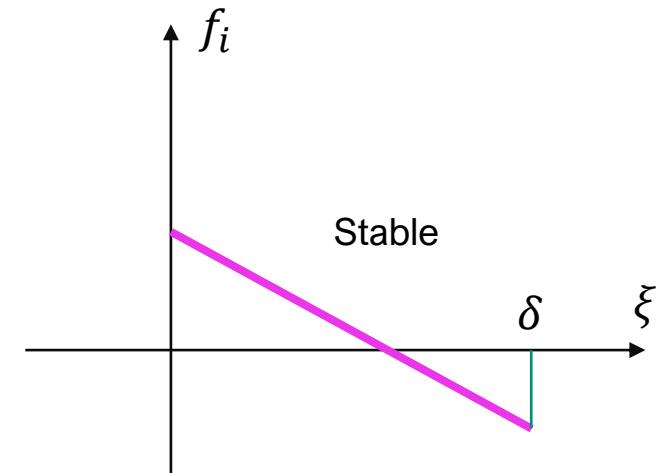
- Assume

$$u_2 = e^{i(kx - \omega t)}.$$

- Repeat derivation of the dispersion curve.

$$\rho\omega^2 = \left[ \frac{-a}{12} \int_0^\delta f_i(\xi) \xi^3 \, d\xi \right] k^4.$$

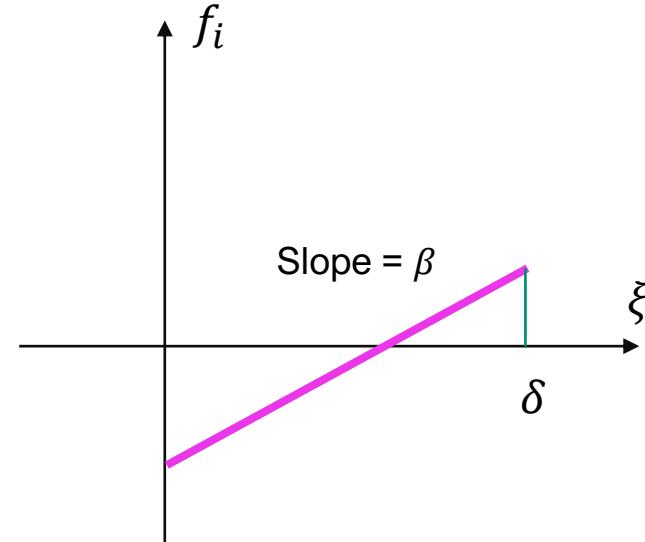
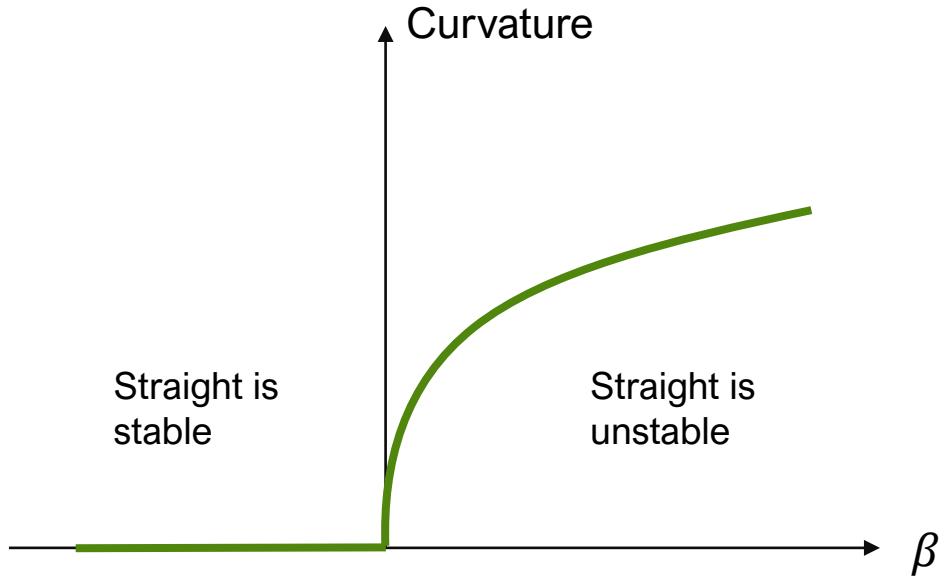
- Long bonds in compression  $\implies \omega$  is real (stable).
- Long bonds in tension  $\implies \omega$  is imaginary (unstable).



# If a straight fiber is unstable, what does equilibrium look like?



- We can compute\* the curvature of a fiber in equilibrium.
- Result:



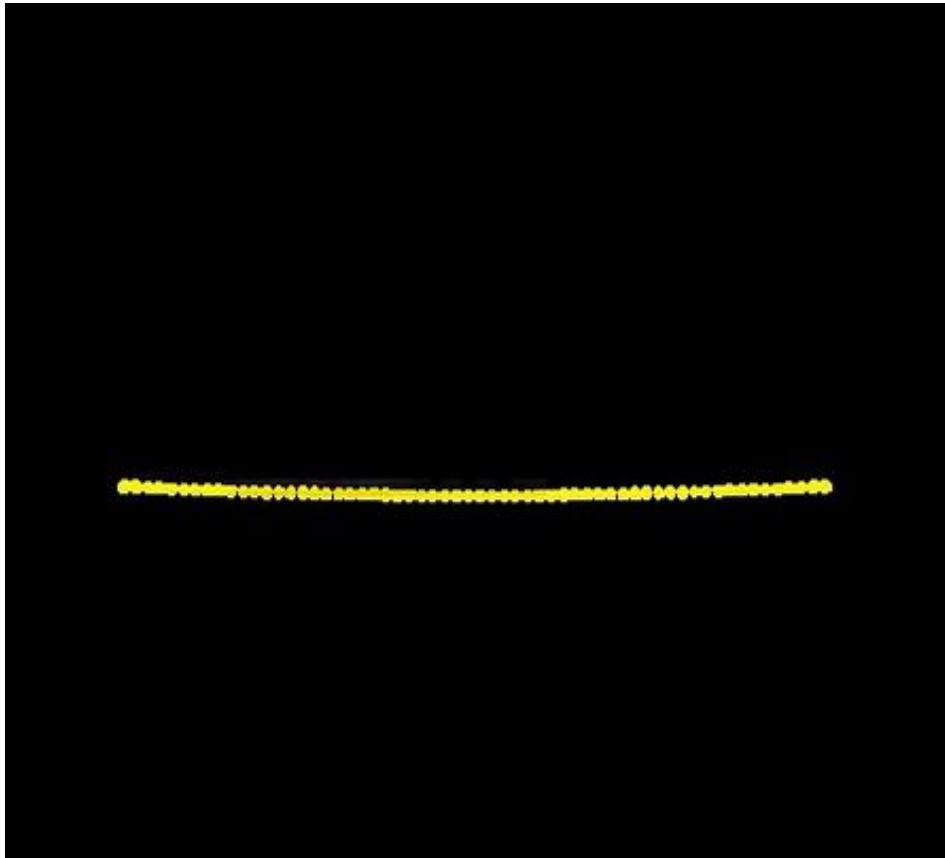
\* SS, "Self-induced curvature in an internally loading peridynamic fiber," technical report SAND2022-5539

# Emu simulation of an internally loaded fiber

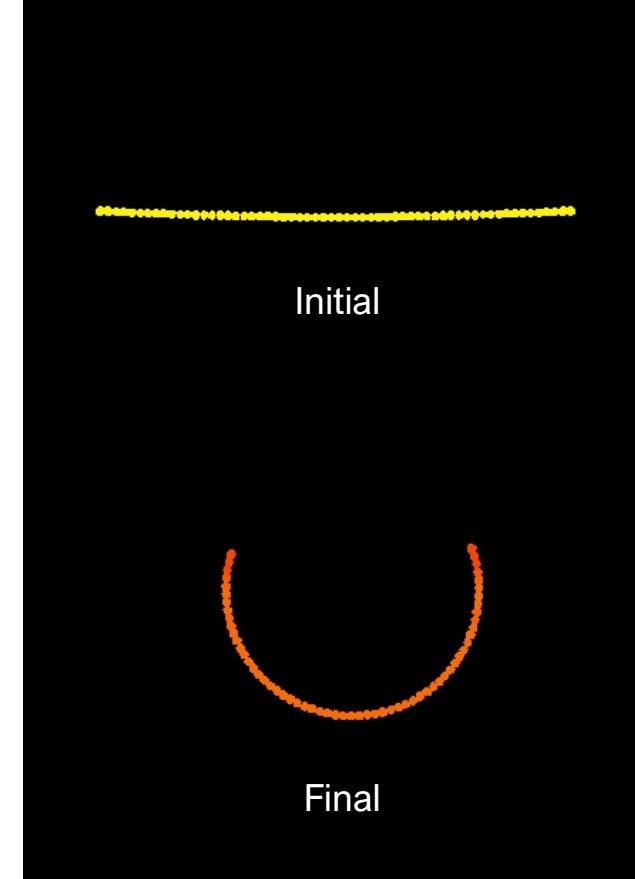


- Internal loading is turned on suddenly.

VIDEO



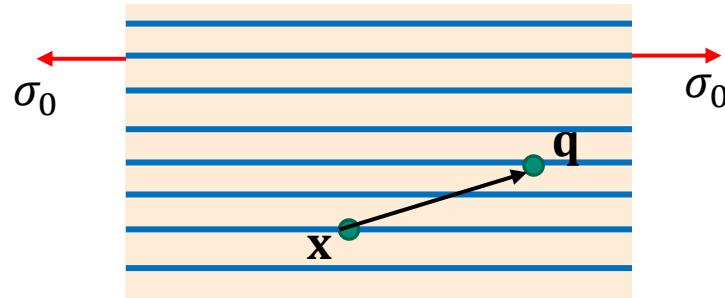
Colors show strain  
(red = 0.01)



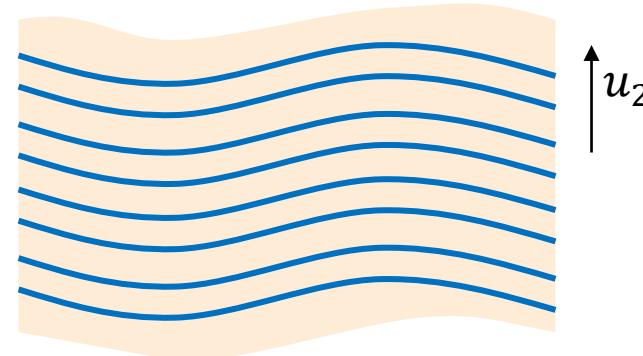
Initial

Final

# Homogenized model of many fibers: Kink bands in a fiber-reinforced composite



Fibers in a matrix  
Remote loading in fibers is  $\sigma_0$   
Bonds can be either stiff or soft



Transverse wave

- The dispersion relation for transverse waves turns out to be

$$\rho\omega^2 = (\sigma_0 + \mu(k))k^2$$

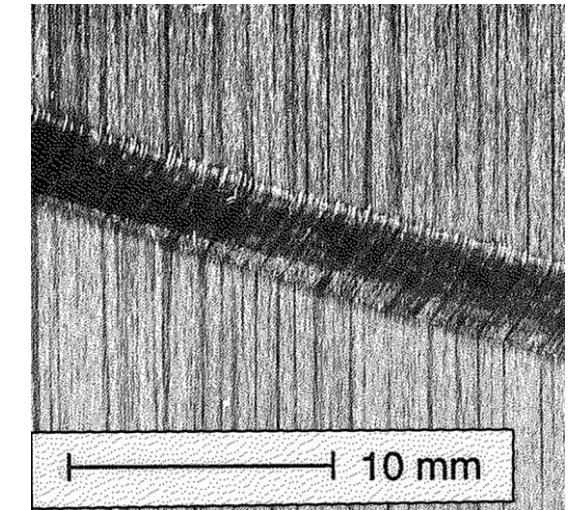
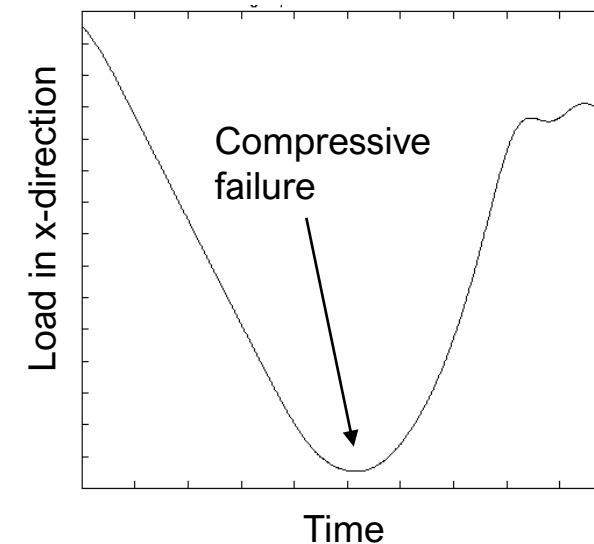
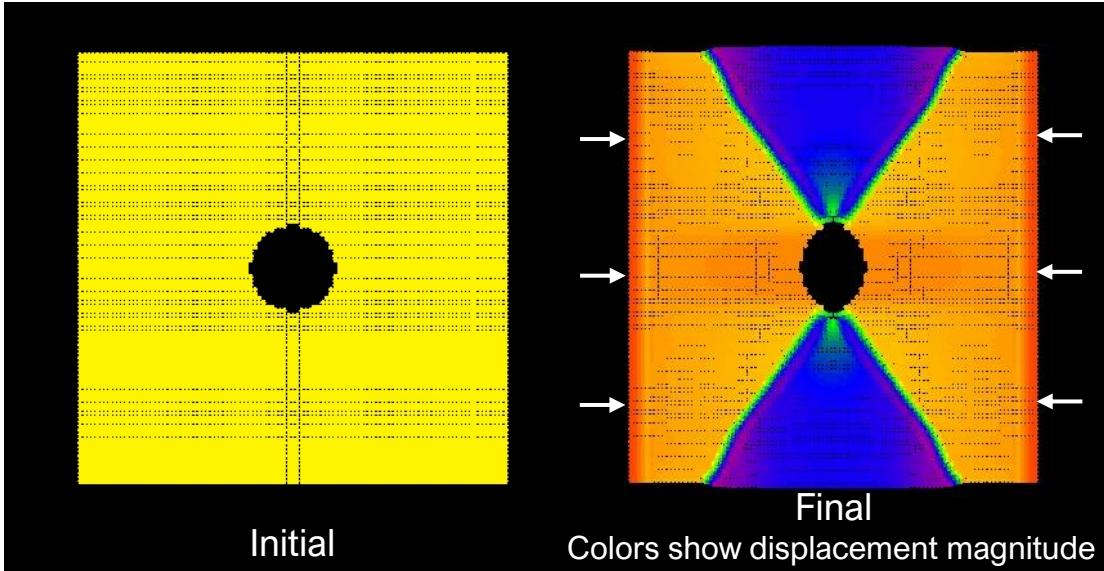
where  $\mu(k)$  characterizes the matrix shear response.

- If the remote loading is compressive ( $\sigma_0 < 0$ ) then  $\omega$  can be imaginary.

# Emu simulation of instability in compression in a composite



- Bonds in the horizontal direction are more stiff than the others.
- Anisotropic, microelastic material model.
- There is no damage (bond breakage) in the peridynamic model.

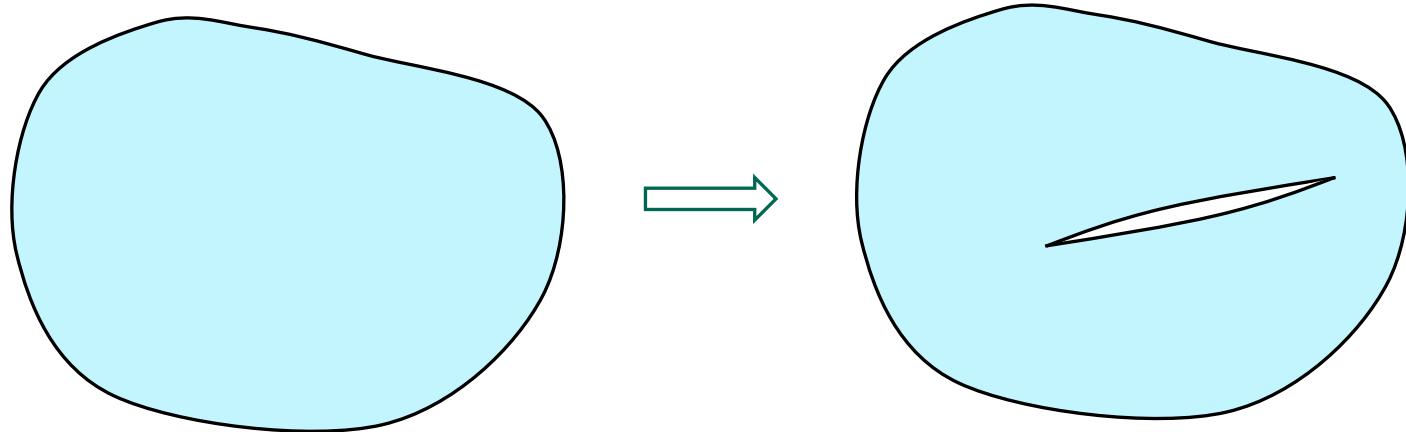


Kink band in a composite  
Image: S.P.H. Skovsgard, thesis, 2019

# Crack nucleation as a material instability



- A body is initially continuous.
- At some later time there is a discontinuity.
- What mathematical conditions need to exist for this to happen?
  - This question is not addressed by fracture mechanics, which assumes a pre-existing defect.

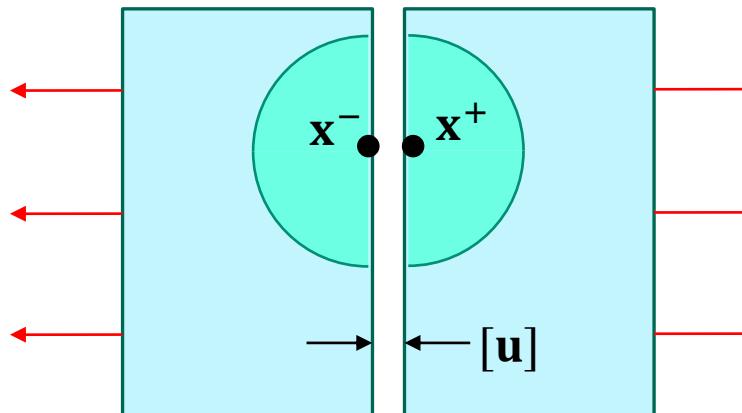
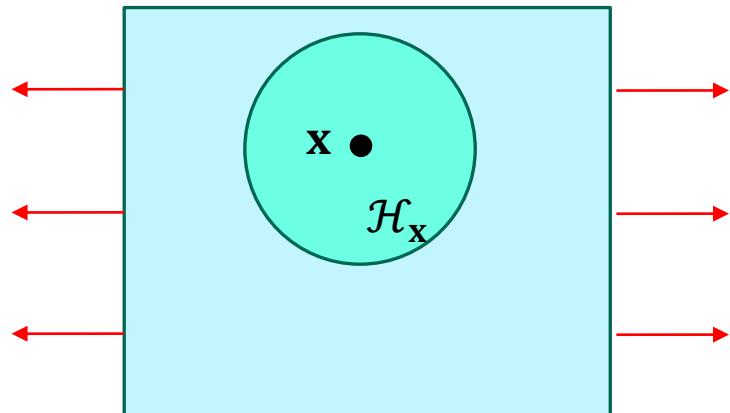


- SS, O. Weckner, E. Askari & F. Bobaru, *Int. J. Fracture* (2010)
- R.P. Lipton, R.B. Lehoucq & P.K. Jha, *J. Peridynamics & Nonlocal Modeling* (2019)

# Perturbation by a jump



- Suppose a small virtual discontinuity is inserted into a body under load.
- Does the discontinuity grow or close up?
- Analyze the evolution of the jump  $[u]$ .



# Perturbation by a jump



- From the momentum balance, find that

$$\rho[\ddot{\mathbf{u}}] = - \left( \int_{\mathcal{H}_x} \mathbf{C}(\xi) d\xi \right) [\mathbf{u}] = -\mathbf{P}[\mathbf{u}].$$

- The gap closes up if

$$[\ddot{\mathbf{u}}] \cdot [\mathbf{u}] < 0.$$

- If

$$\mathbf{u}_0 \cdot (\mathbf{P}\mathbf{u}_0) > 0 \quad \forall \mathbf{u}_0 \neq 0$$

then a crack cannot form.

- Let  $p_1, p_2, p_3$  be the eigenvalues of  $\mathbf{P}$  (which is symmetric).
- Stability index:*

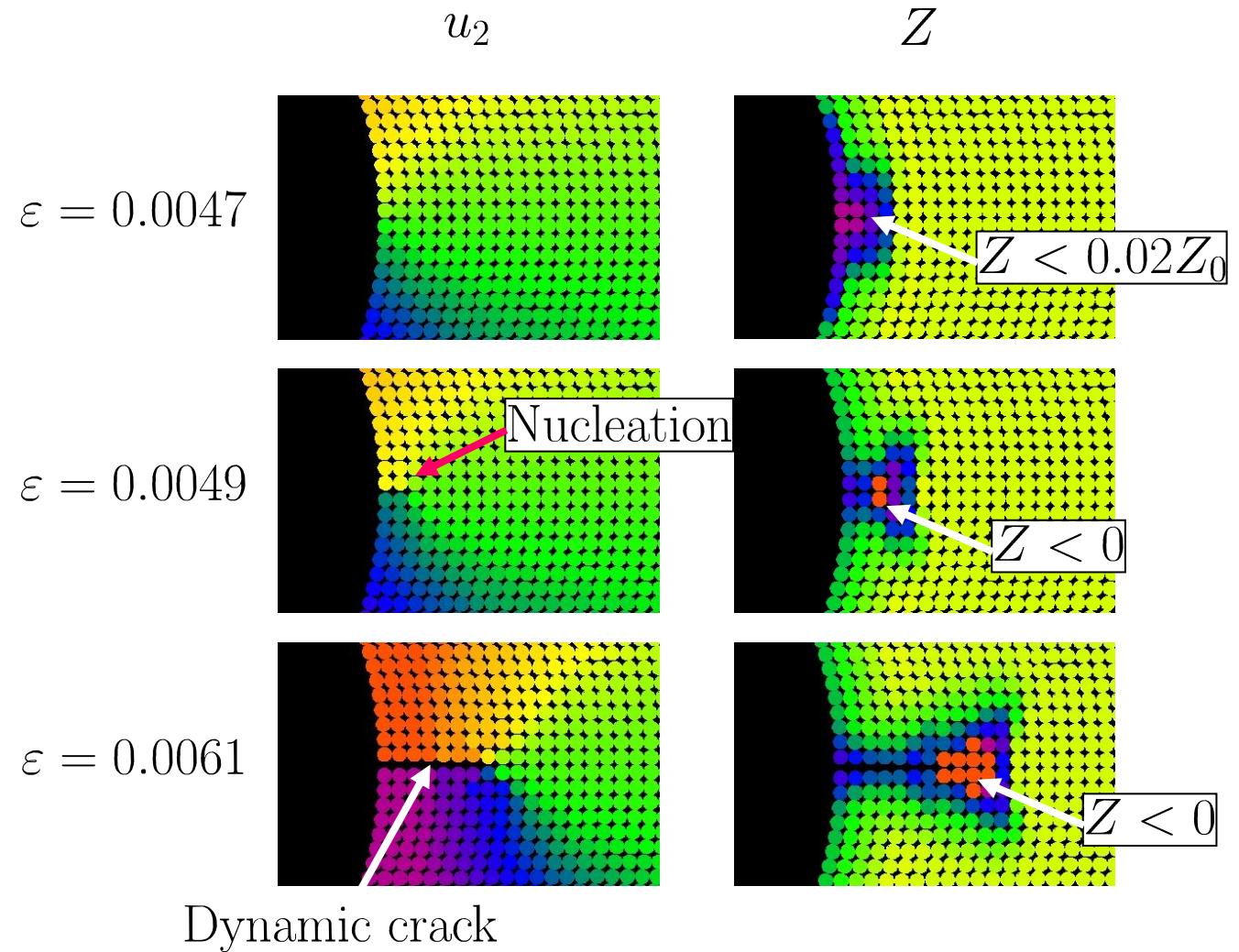
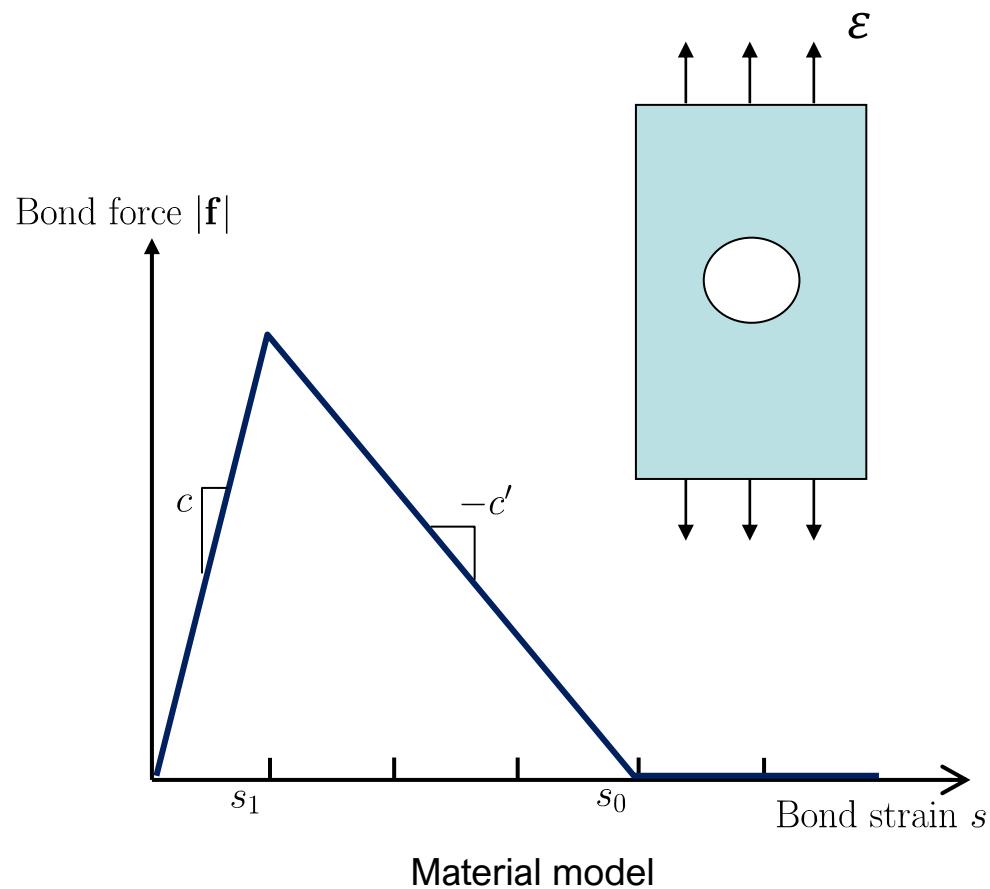
$$Z = \min\{p_1, p_2, p_3\}.$$

- If  $Z \leq 0$ , a crack can nucleate. The eigenvector gives the gap displacement.
- The orientation of the crack comes from the underlying stress field.

# $Z < 0$ at the crack nucleation site and near a growing crack tip



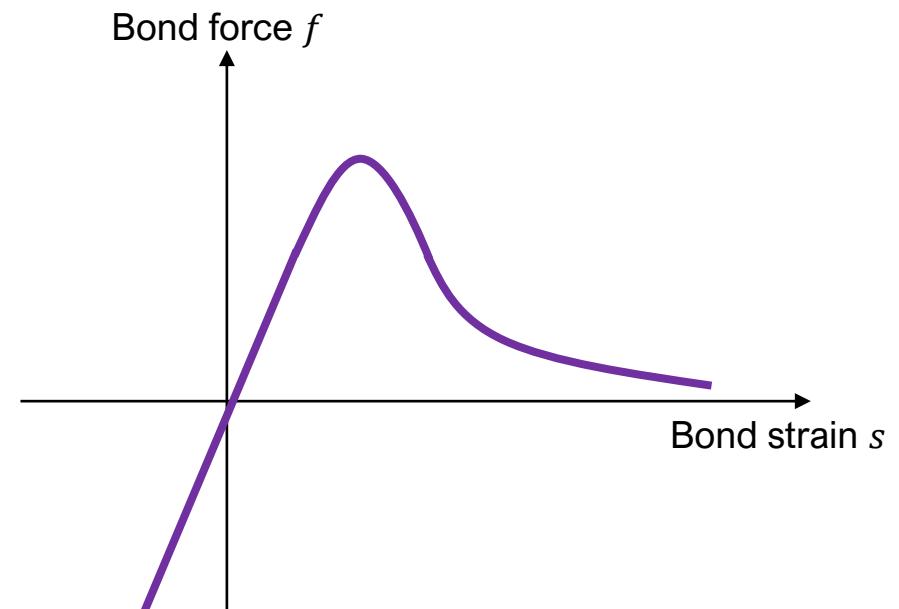
- There is a process zone near a growing crack tip within which  $Z < 0$ .



# A nonlocal model of fracture can be well-posed (!)



- Nonconvex elastic peridynamic material.
- Points entering the unstable branch lead to creation of a jump in displacement.
- This discontinuity can grow through the body.
- As it grows, it consumes a definite amount of potential energy per unit crack area (Griffith crack).
- The whole problem can be well-posed.
- These results hold for both static and dynamic cases.

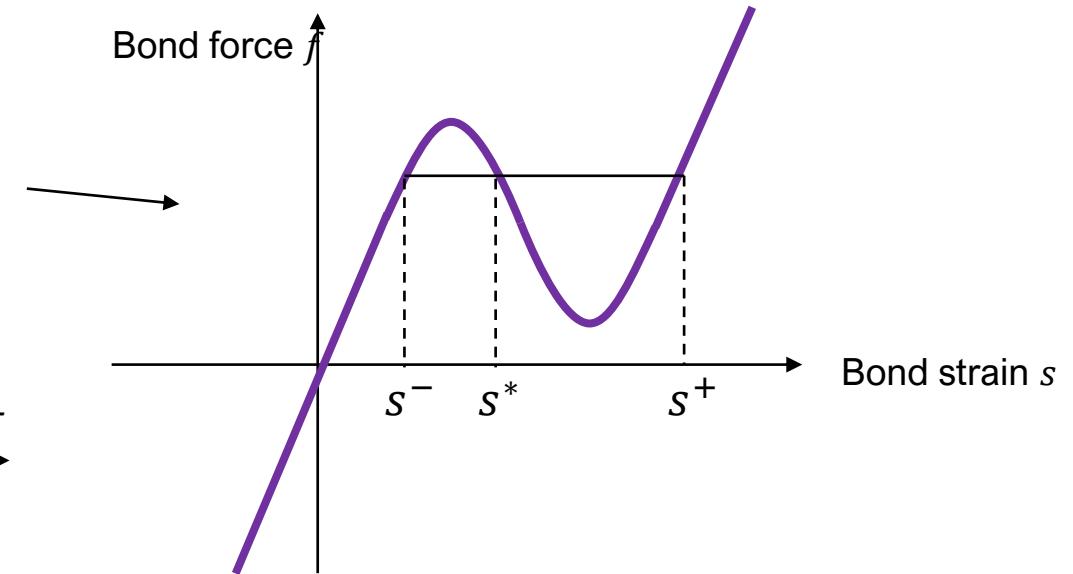
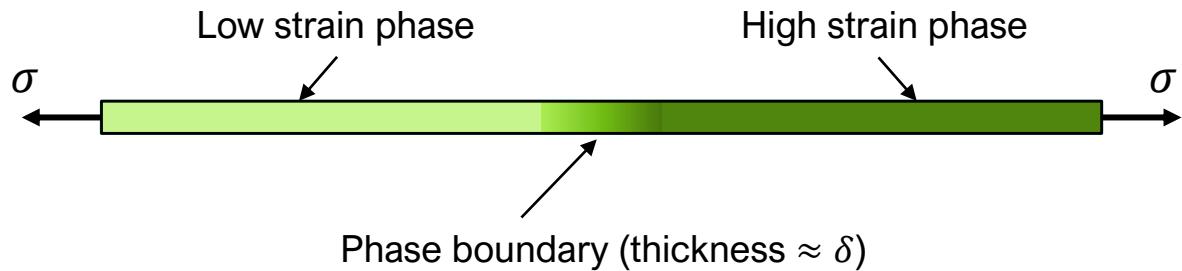


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# Mechanically induced phase transformation



- Here's another type of nonconvex microelastic material:



- There are multiple bond strains for the same bond force.
- This creates the possibility of multiple phases in a bar in equilibrium.
- Within the phase boundary, there are always some bonds in the unstable part of the material model.

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# What conditions permit coexistent phases?

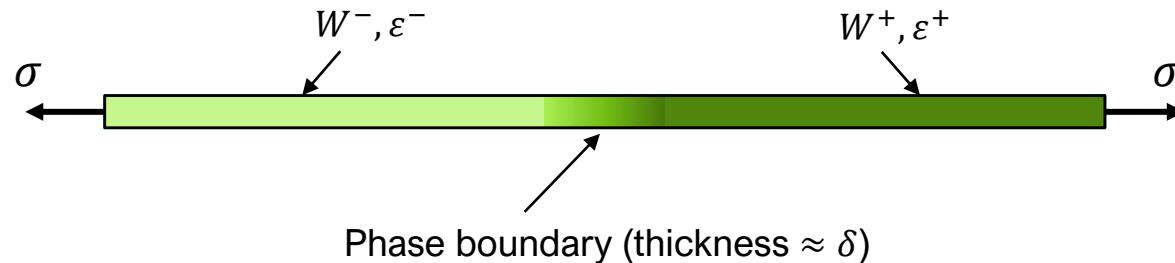


- In equilibrium, energy conservation implies

$$W^+ - W^- = (\varepsilon^+ - \varepsilon^-)\sigma$$

which is formally the same as in the local theory.

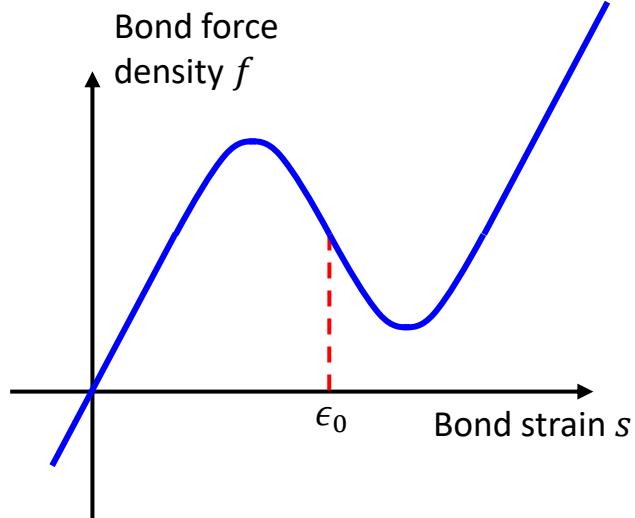
- Compare Weierstrass corner condition, Maxwell condition in the calculus of variations.



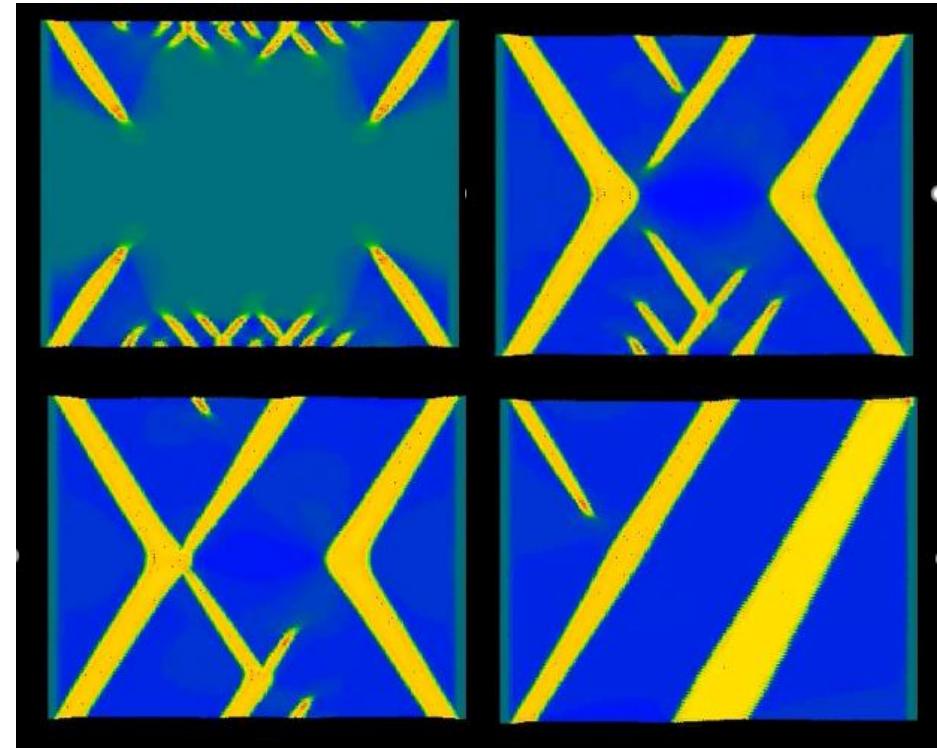
# In peridynamics, a phase boundary contains potential energy



- The system evolves over time to reduce the total surface area of phase boundaries.
- We can use this to simulate microstructure evolution.



Microstructure evolution in a plate with initial strain  $\epsilon_0$   
Colors show bond strain





VIDEO

Microstructure evolution in a plate with initial strain  $\varepsilon_0$   
Colors show bond strain





- Nonlocality changes everything about material stability.
- We can do meaningful continuum mechanics within an unstable material.
- We can use this to simulate real phenomena:
  - Rate effect on bulk material strength
  - Fracture nucleation, Griffith criterion
  - Compressive failure modes in composites
  - Microstructure evolution