



The role of pressure-dependent viscoplasticity and volumetric dilatation in energetic materials at intermediate strain rates



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Mechanical Behavior of Energetics: A Modeling Challenge



- Why care about mechanical behavior? –Safety is key
- Precursor to hot spot formation and High Explosive Violent Reaction (HEVR) events

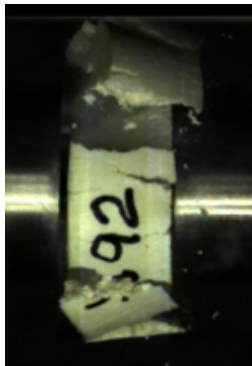
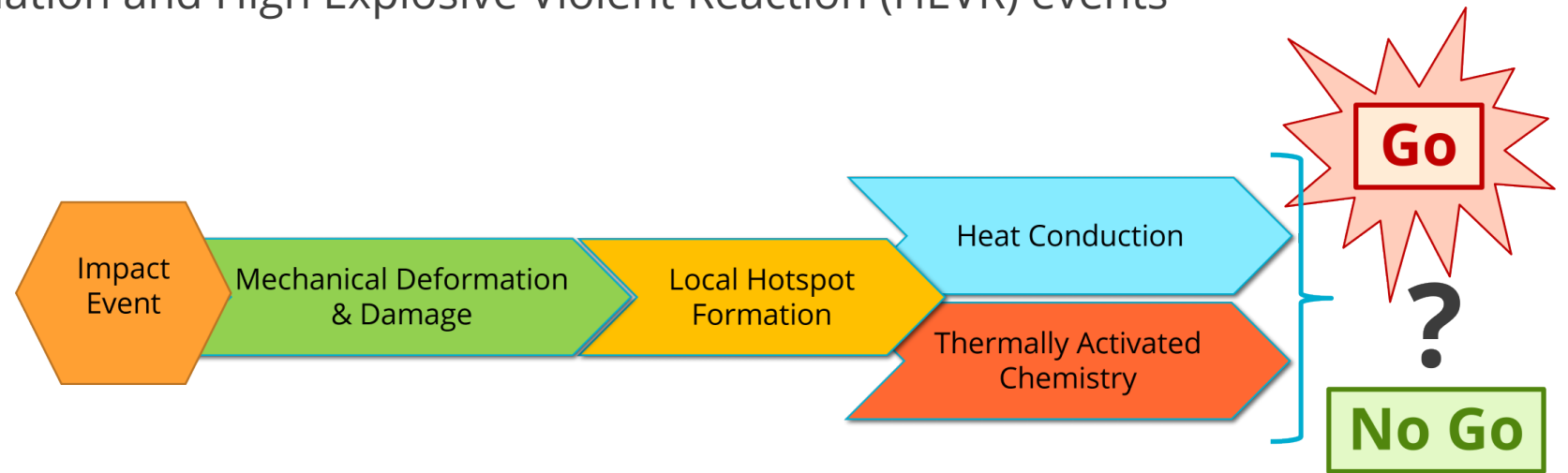


Image: courtesy Marcia Cooper

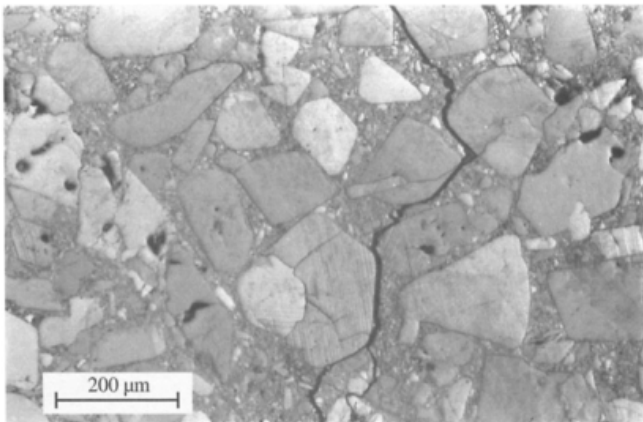


- What is behavior of a mechanically deformed explosive exposed to other environments?—must first be able to predict the deformed state

PBX Mechanical Behavior: A Modeling Challenge



- PBX Mechanical behavior is complex:
 - Strain rate dependent
 - Temperature dependent
 - Pressure dependence
 - Tension-Compression Asymmetry
 - Many inelastic deformation mechanisms: Viscoelasticity (binders), Cracking (intra- and inter-granular), Porosity opening, dislocation slip, twinning (some energetic crystals)



Plastic Bonded Explosive [Rae, 2002]

How to represent various inelastic mechanisms in a macroscale model?

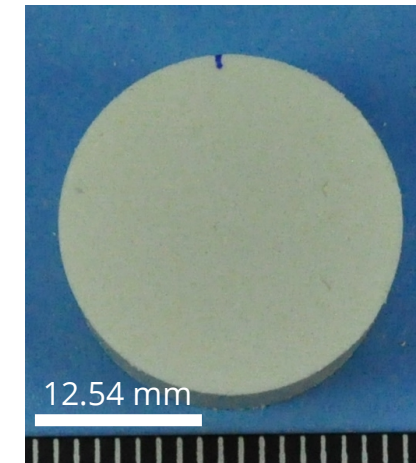
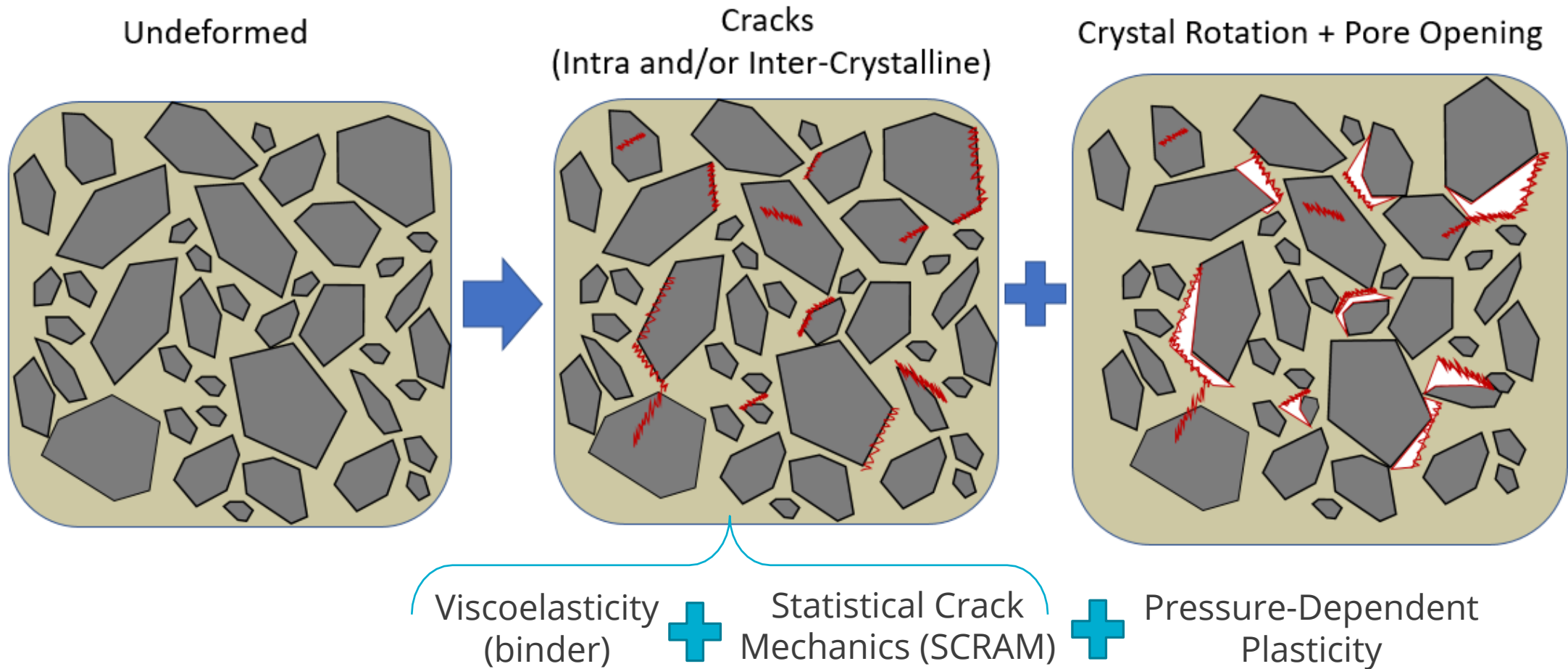


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Hypothesized Mesoscale Deformation Mechanisms



- Simplified view of mesoscale processes and macroscale interpretation



➤ Kinematics:

$$\epsilon = e + \frac{1}{3} \epsilon_{\text{vol}} \mathbf{I}$$

$$\sigma_m = K \epsilon_{\text{vol}}$$

$$e = (e^{ve} + e^D) + e^p$$

➤ Viscoelasticity

$$\dot{s} = 2G^\infty \dot{e}^{ve} + \sum_{\kappa=1}^N \left(2G^{(\kappa)} \dot{e}^{ve} - \frac{s^{(\kappa)}}{\tau^{(\kappa)}} \right)$$

Prony series of shear moduli and relaxation times

$$\dot{s}^{(\kappa)} = 2G^{(\kappa)}(\dot{e} - \dot{e}^p) - \frac{s^{(\kappa)}}{\tau^{(\kappa)}} - \frac{G^{(\kappa)}}{G_0} \left[\frac{3}{a} \left(\frac{c}{a} \right)^2 \dot{c} s + \left(\frac{c}{a} \right)^3 \dot{s} \right]$$

➤ SCRAM Damage

$$e^D = \frac{1}{2G_0} \left(\frac{c}{a} \right)^3 s$$

$$\dot{c} = \begin{cases} v_{res} \left(\frac{K_I}{K_1} \right)^m & \text{for } K_I < K' \\ v_{res} \left[1 - \left(\frac{K_{0\mu}}{K_I} \right)^2 \right] & \text{otherwise} \end{cases}$$

$$D = \frac{\left(\frac{c}{a} \right)^3}{1 + \left(\frac{c}{a} \right)^3}$$

➤ Drucker-Prager Plasticity

$$f(\sigma_{ij}) = \sigma_e + A \cdot \sigma_m - \sigma_y$$

$$g(\sigma_{ij}) = \sigma_e + B \cdot \sigma_m - \sigma_y$$

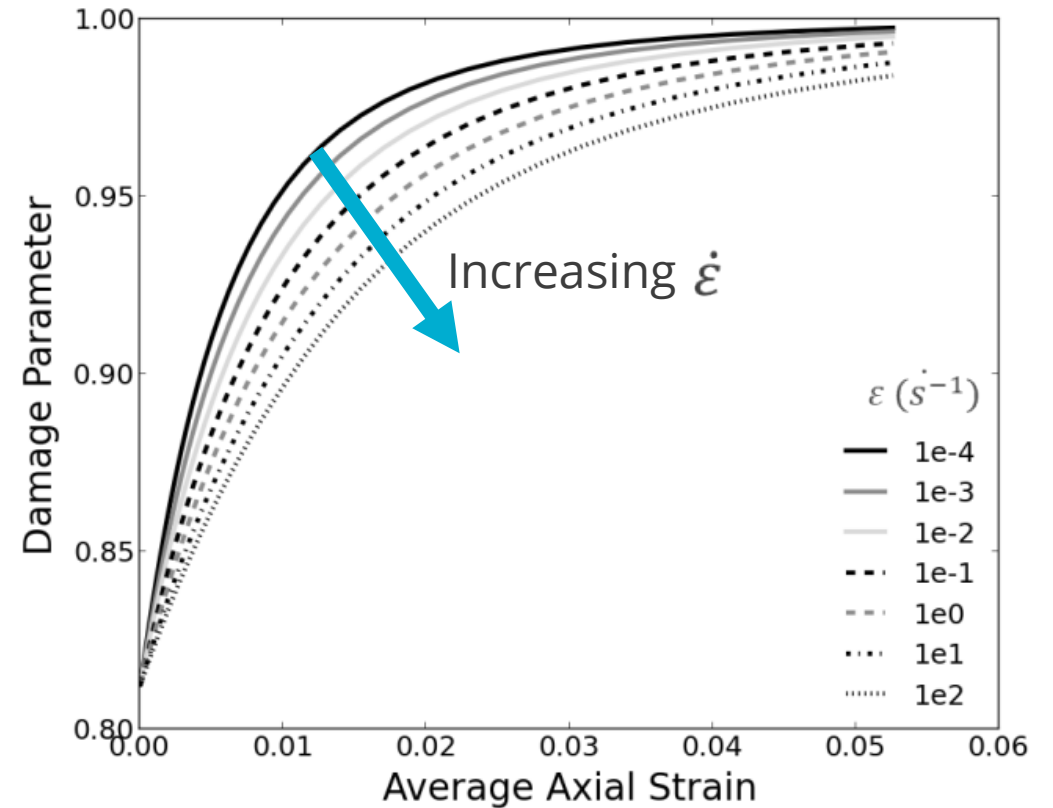
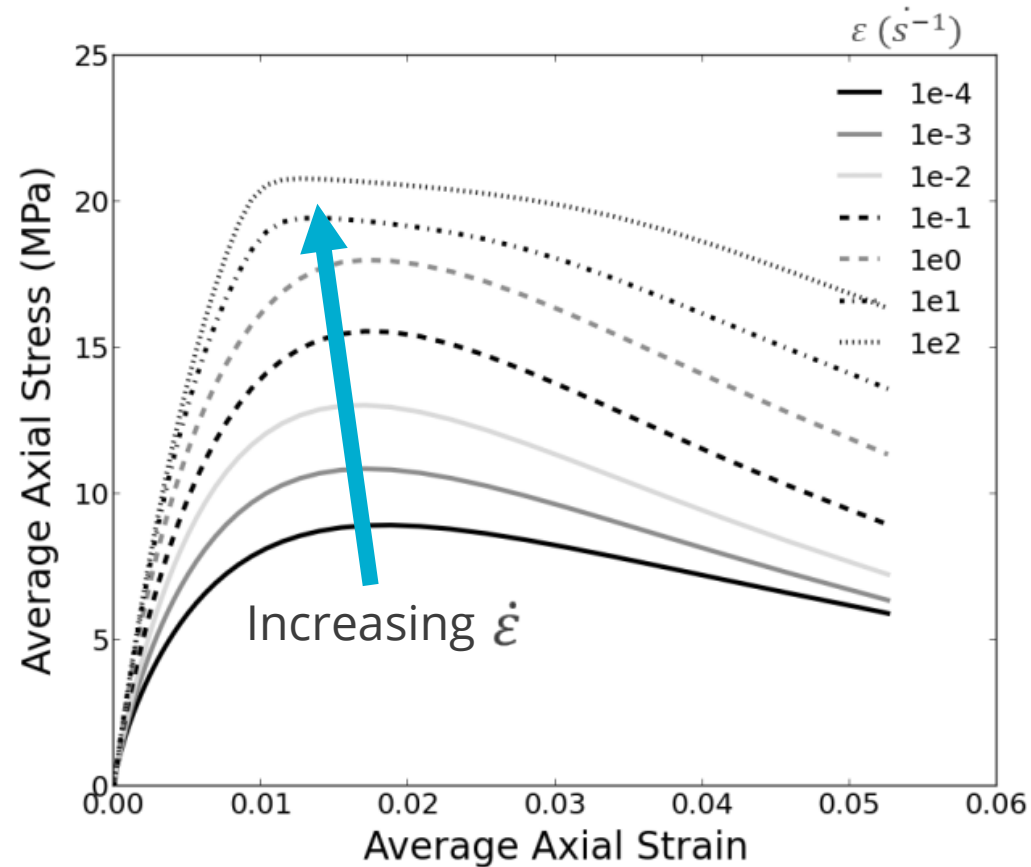
$$\dot{\epsilon}^p = \dot{\lambda} \frac{\partial g}{\partial \sigma}$$

$$\dot{\lambda} = \frac{1}{\tilde{\tau}} \left\langle \frac{f(\sigma_{ij})}{\sigma_0} \right\rangle_{\tilde{m}} \quad \sigma_e = \sqrt{\frac{3}{2} s_{ij} s_{ij}}$$

Uniaxial Compression



- Rate Dependence from Viscoelasticity and SCRAM mechanisms

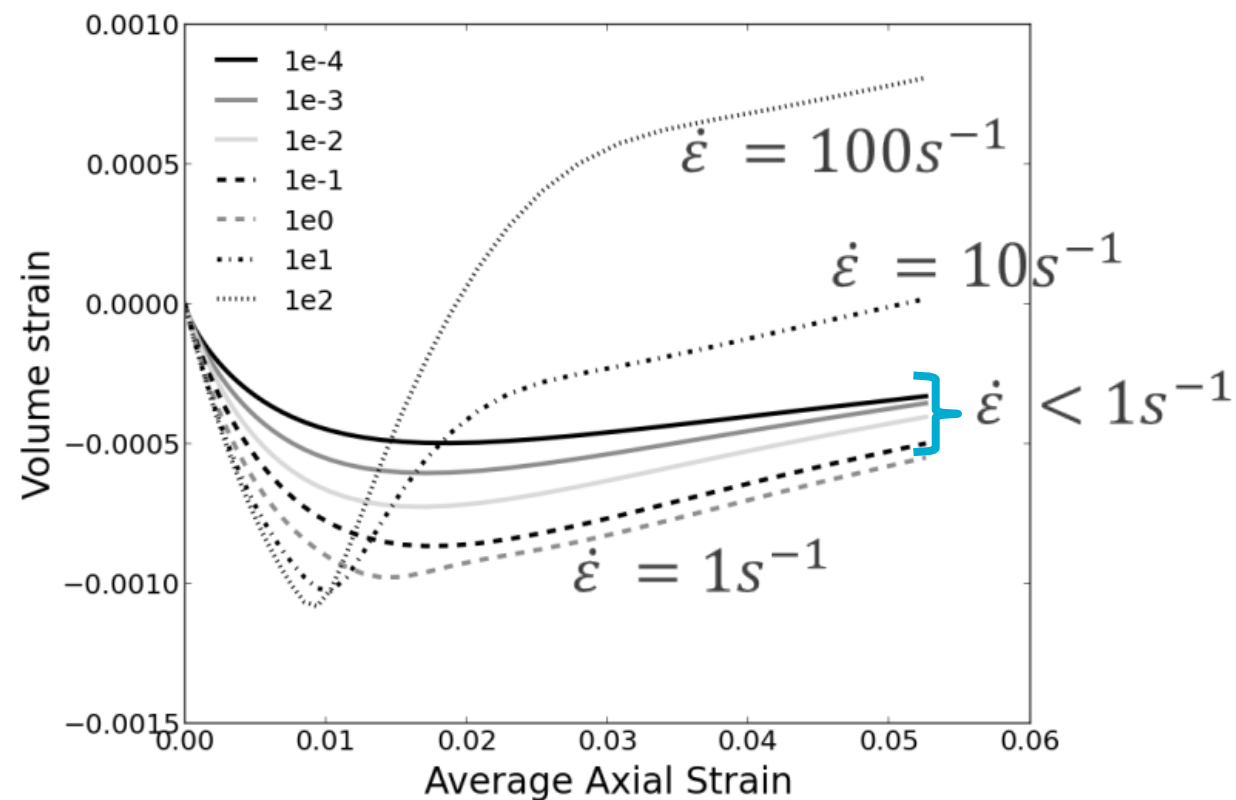
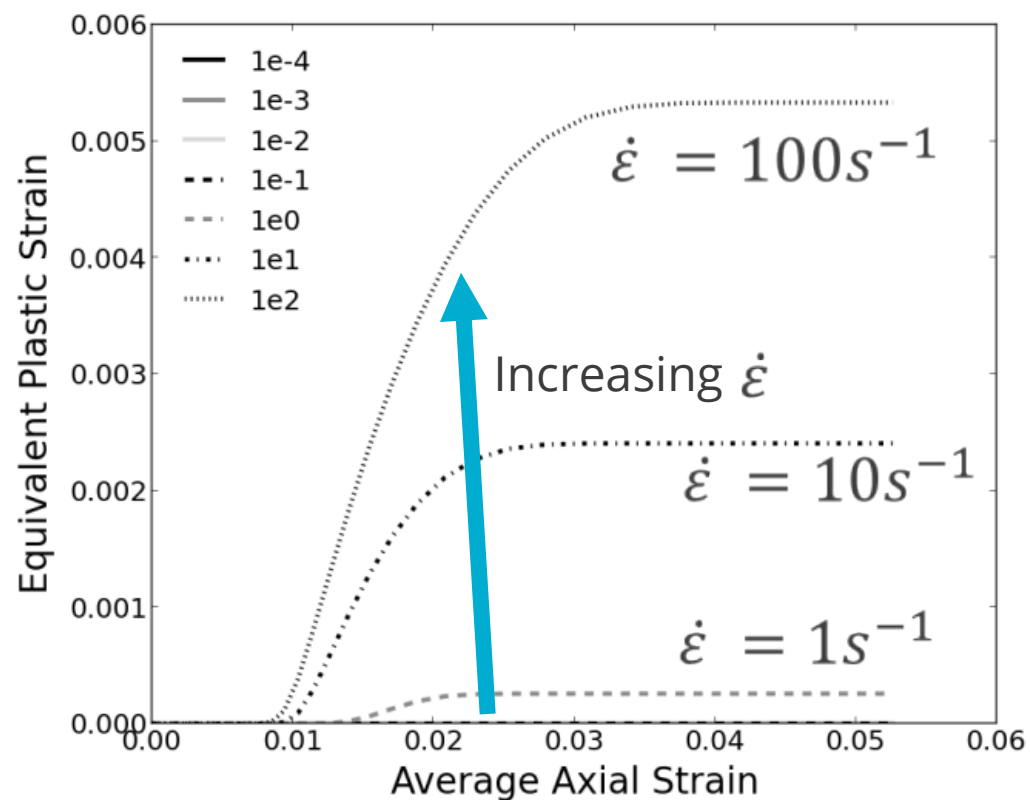


While increasing strain rate increases stiffness from the viscoelasticity, the damage evolution is reduced with increasing strain rate. Why?

Uniaxial Compression

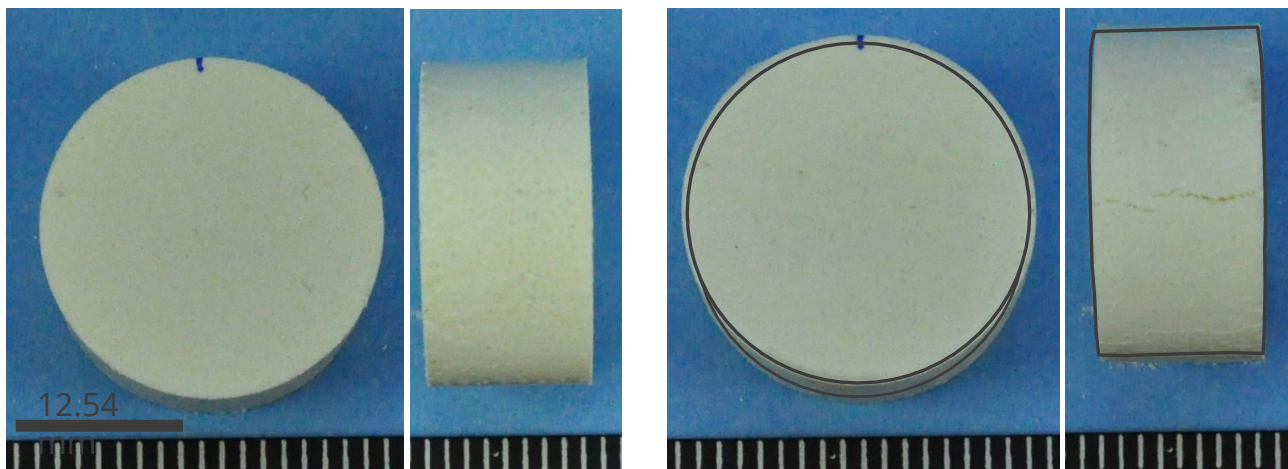
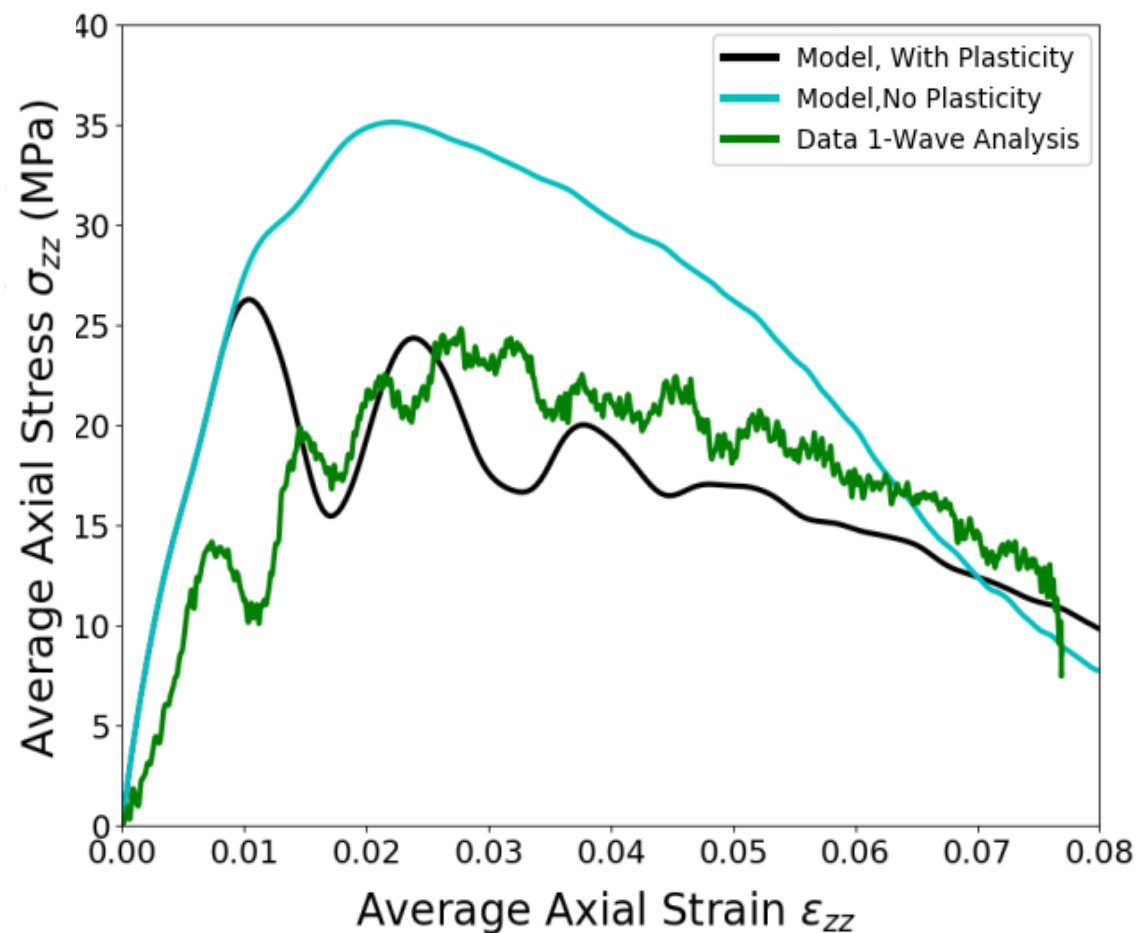
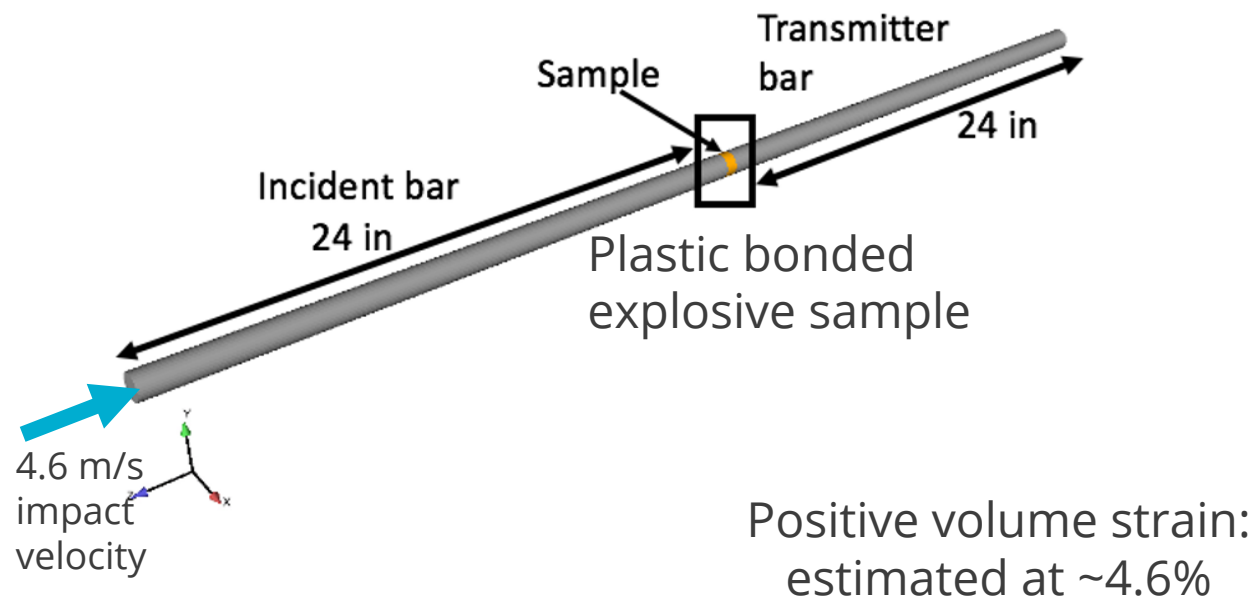


- Plasticity is a second inelastic mechanism, which is only active at strain rates $\dot{\epsilon} \leq 1s^{-1}$



Plasticity limits stress development in the material which reduces the rate of damage and also produces positive volume strain (dilatation)

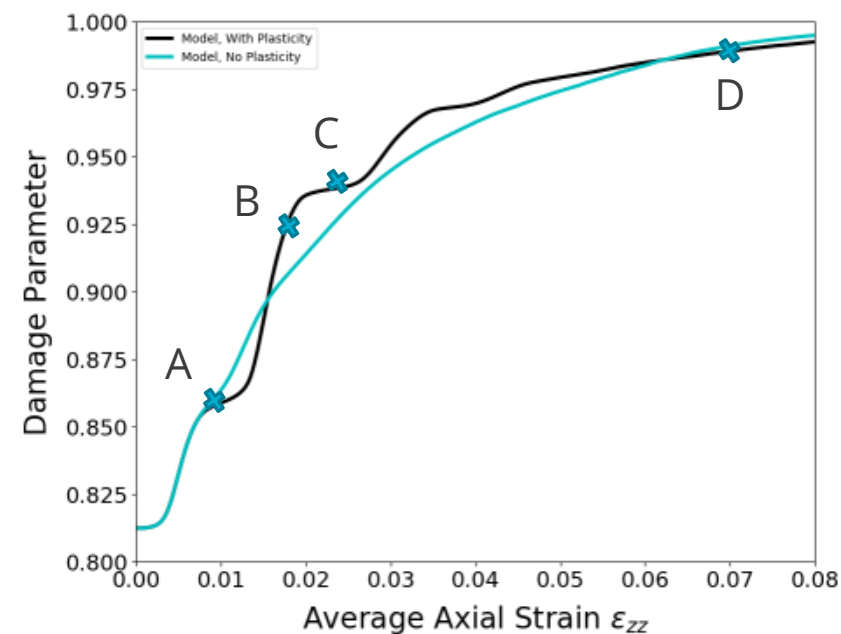
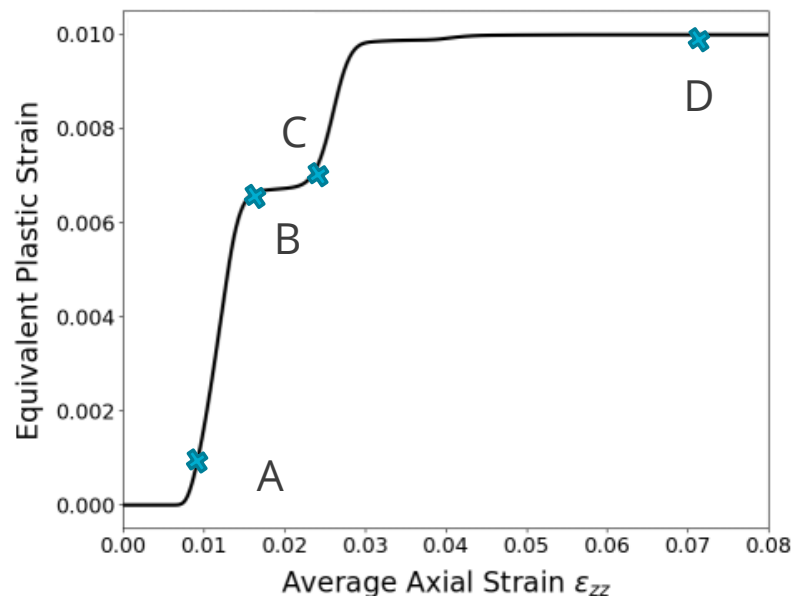
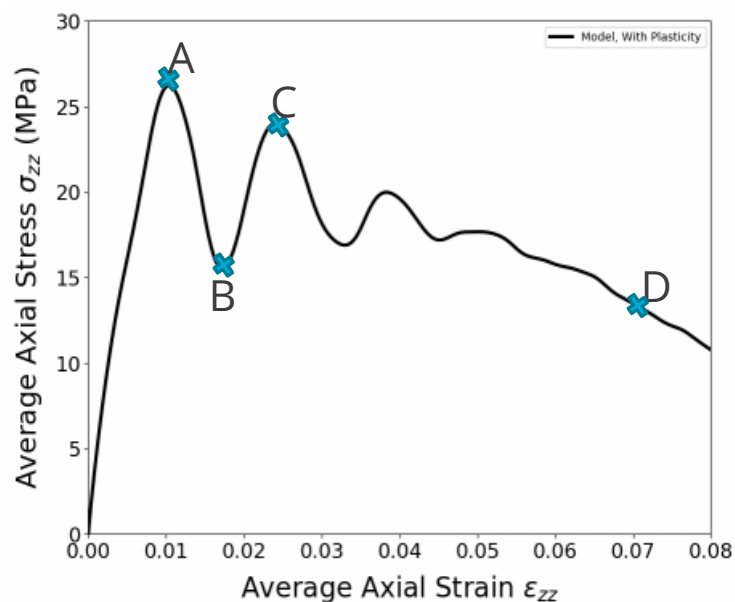
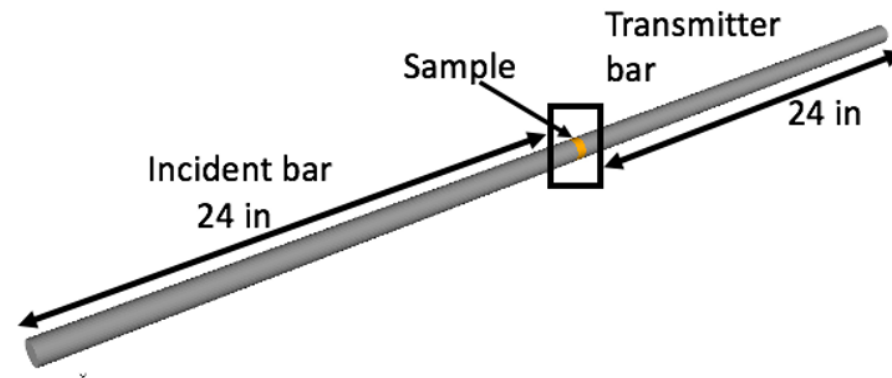
Intermediate Strain Rate Impact



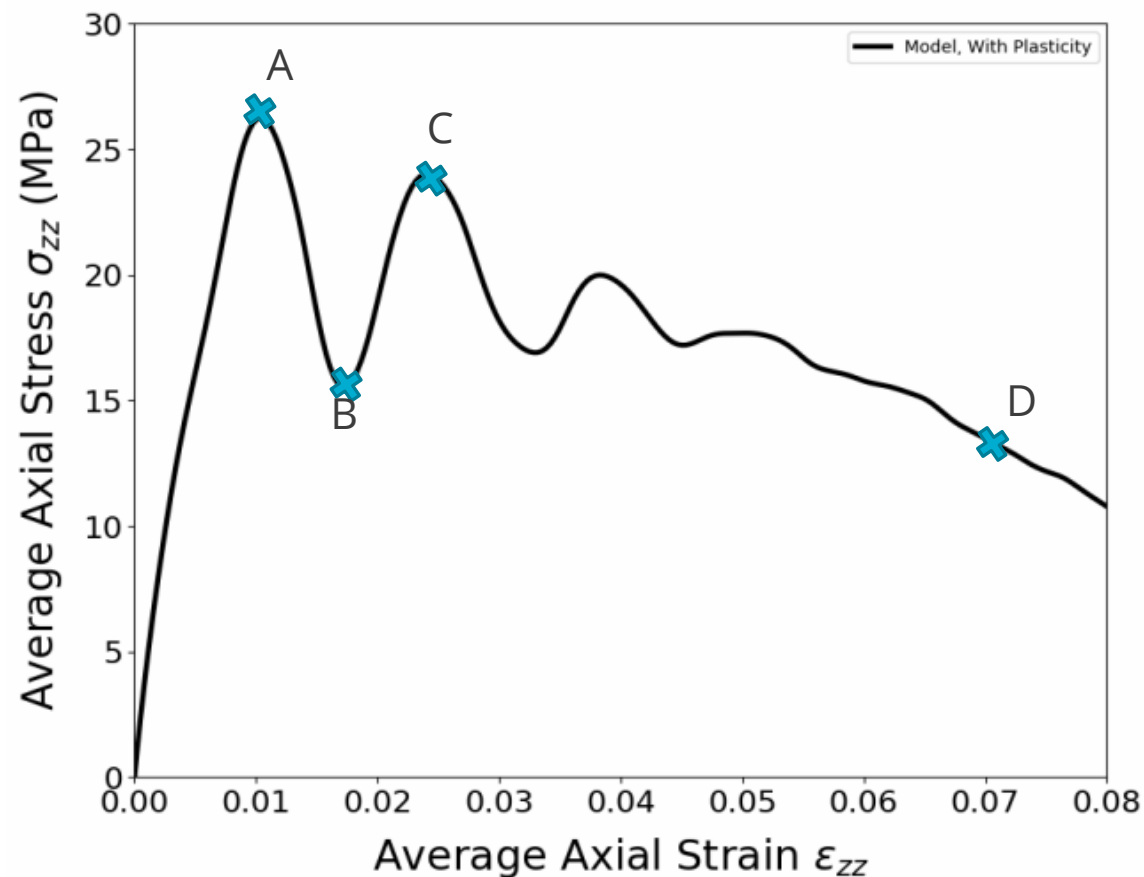
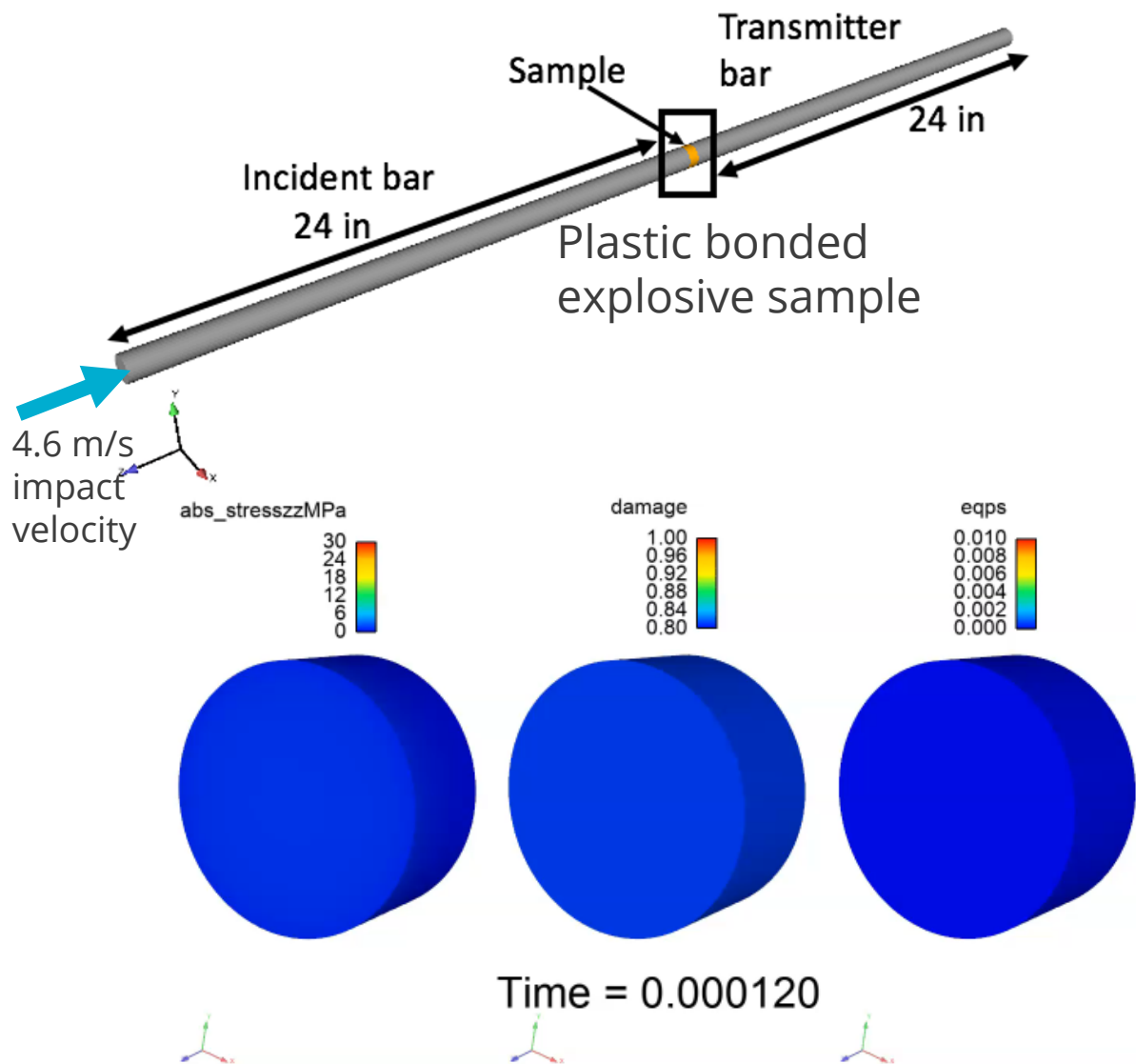
Evolution of SCRAM Damage and Plasticity



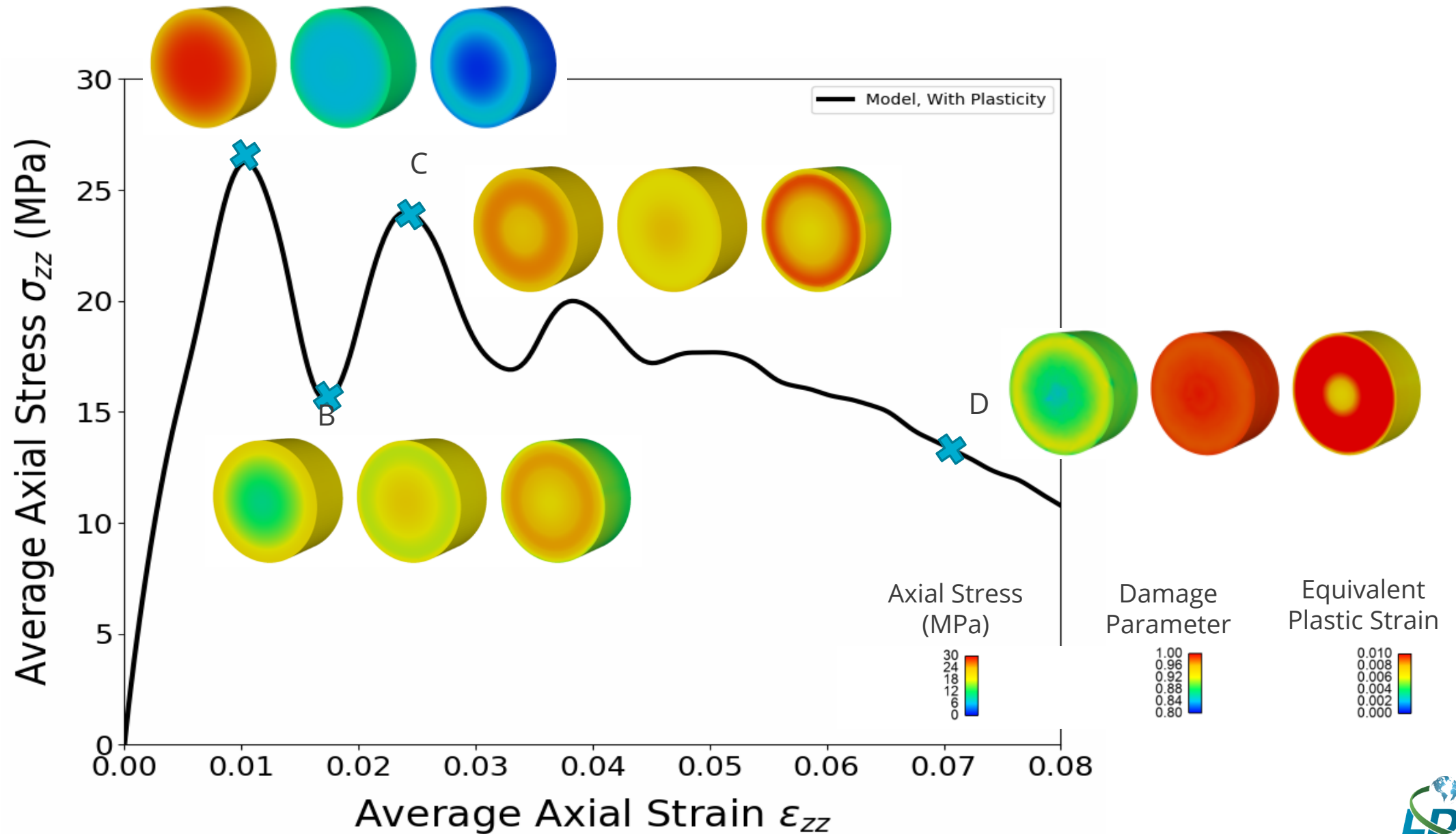
- Both SCRAM damage and plasticity are active simultaneously
- Final damage state is the same, but different evolution behavior
- Plastic flow occurs during stress wave oscillations



Evolution of SCRAM Damage and Plasticity

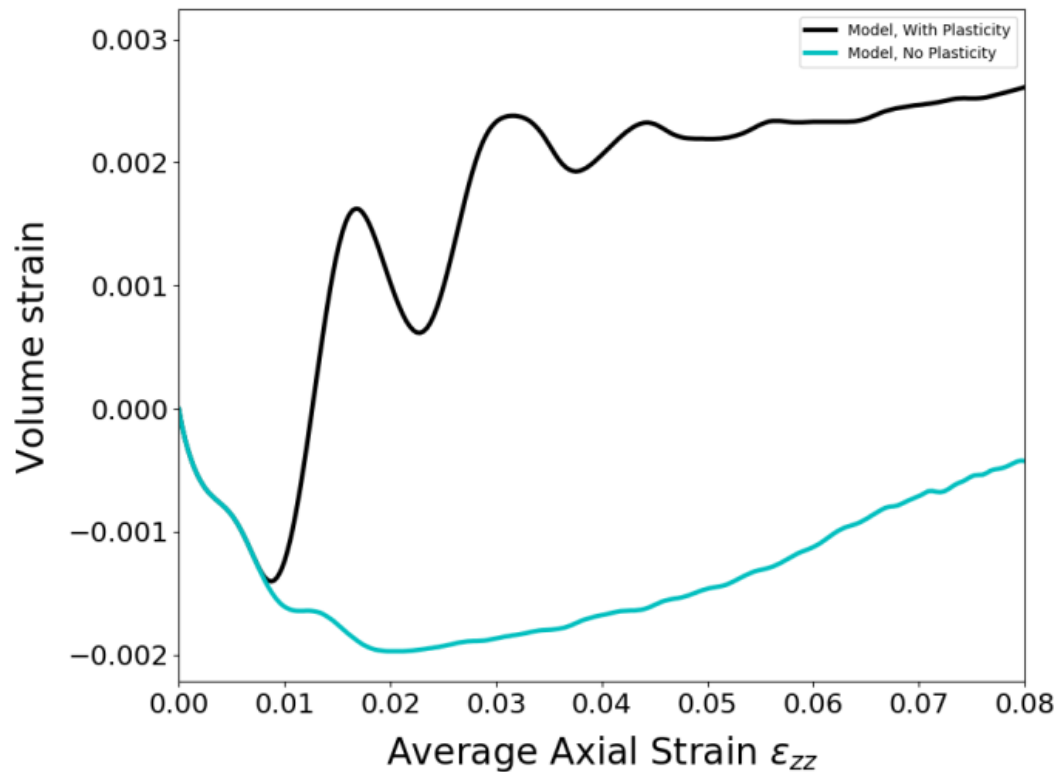


Evolution of SCRAM Damage and Plasticity

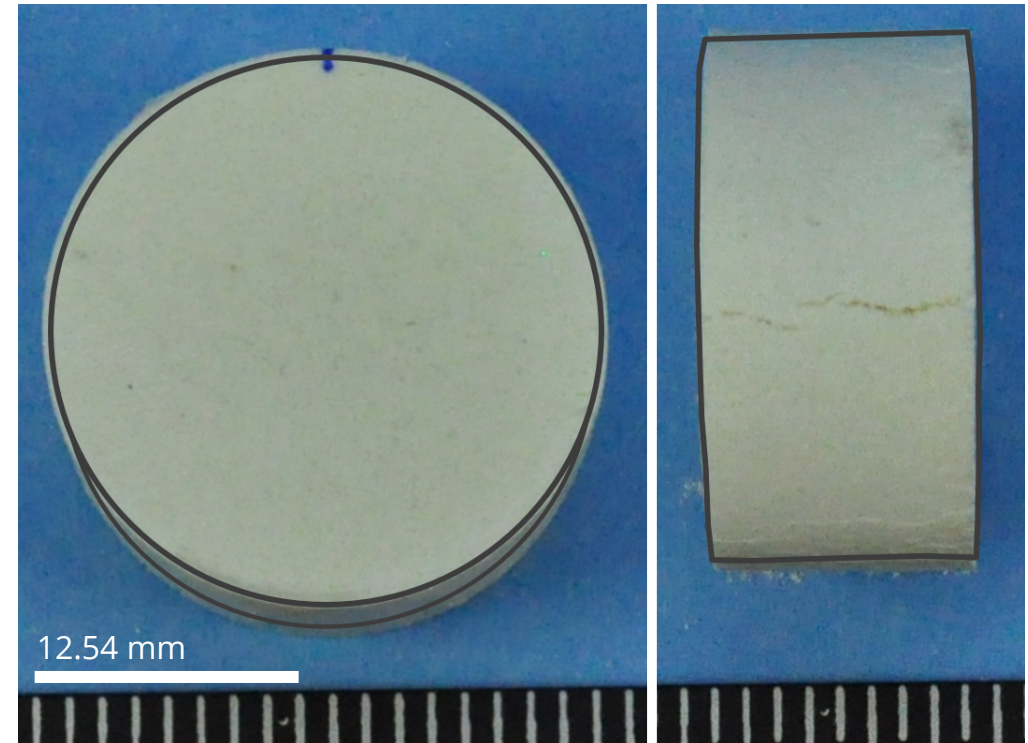


Plasticity allows predictions for net dilatation

- Net dilatation only predicted from model with plasticity
- Volume strain estimated from image analysis of pre- and post-mortem SHPB sample



Positive volume strain: estimated ~4.6%



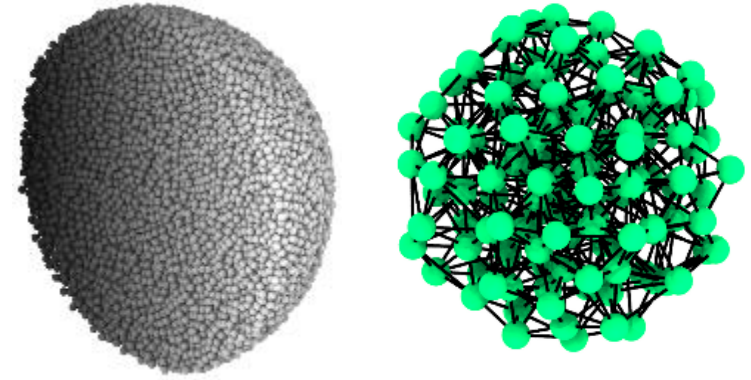
Meshfree Mesoscale Models

Give insight to mechanical deformation and damage mechanisms under many different stress states

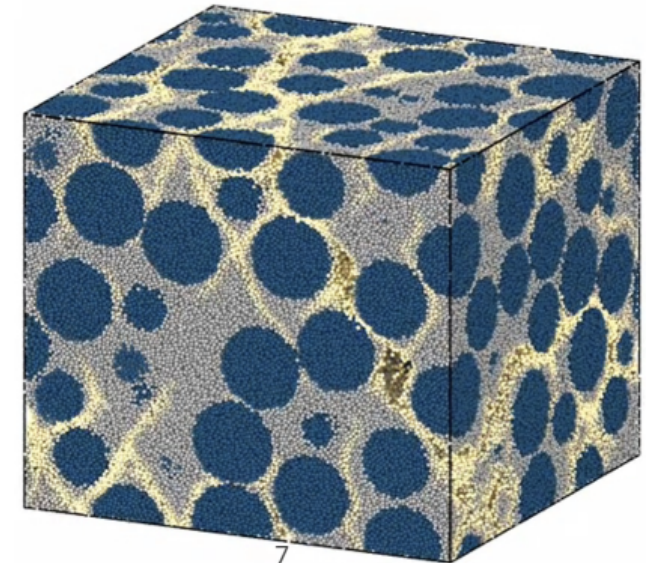
Bonded Particle Models (BPMs)

- BPMs are minimalistic particle-based models for fragmentation – ideal for studying trends/testing theories
- Very efficient, can simulate large systems, $\sim 10^3$ grains, at high resolutions, $\sim 10^4$ particles/grain
- In BPMs, grain = collection of repulsive particles connected by network of (typically) pairwise bonds
- Bonds break under specific criteria – e.g. stretch threshold
- Functional form of bond controls material properties: moduli, fracture toughness, plasticity, viscoelasticity, ...
- Open-sourced models available in recently released LAMMPS package

Grain consisting of $\sim 10^2$ particles

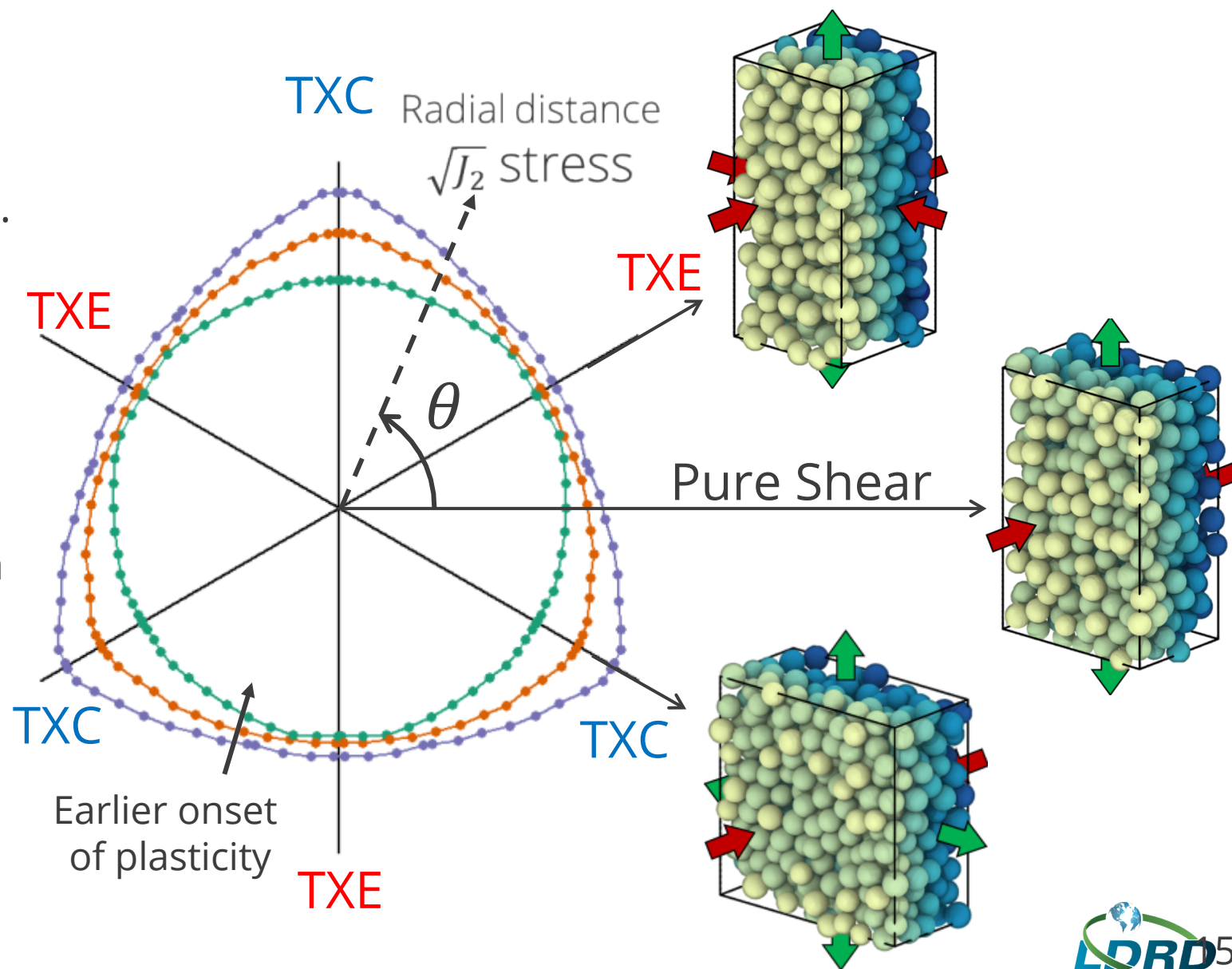
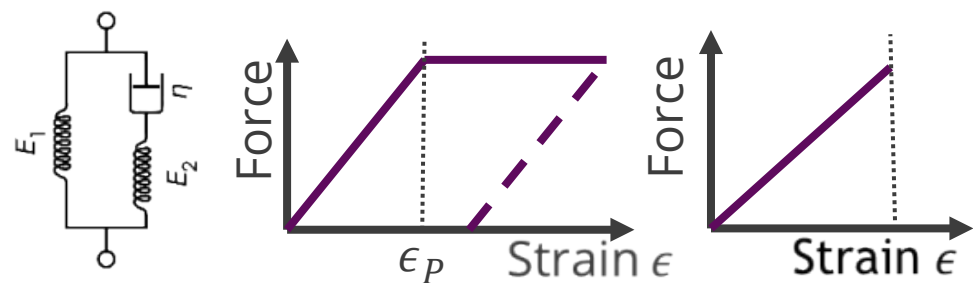


PBX-like composite



Identifying yield surface shape to inform continuum models

- Many possible loading paths, all have unique failure stress. Map out yield surfaces, often assumed to have simple shape (e.g. Drucker-Prager)
- **Testing how changes in binder's material properties impact yield surface**
 - better understanding of inelastic yielding for continuum models

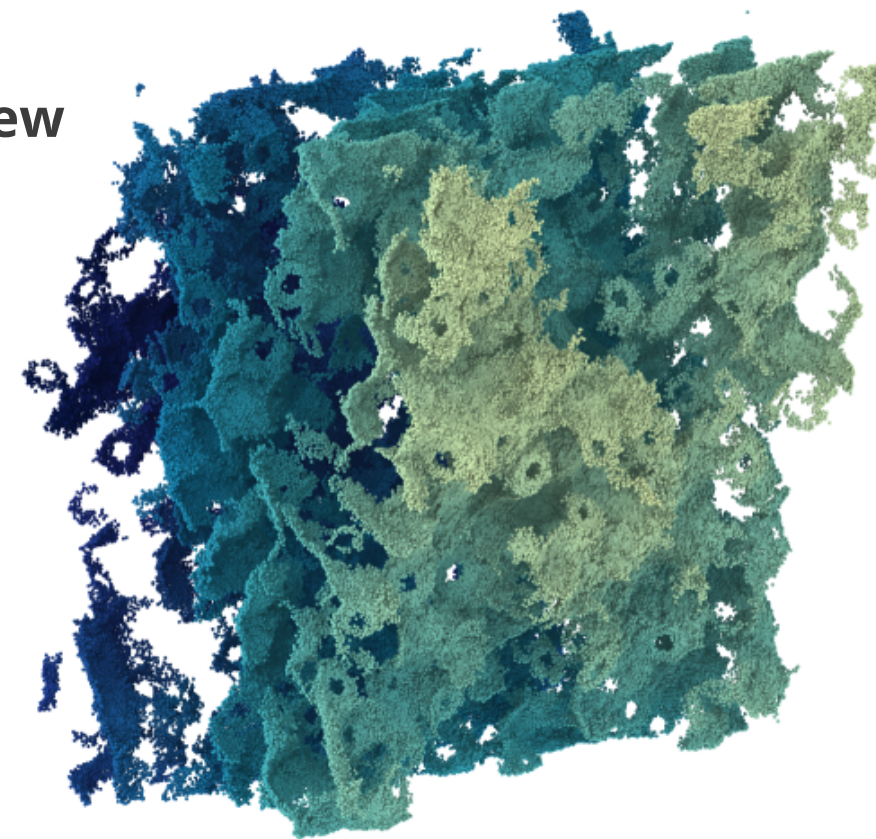
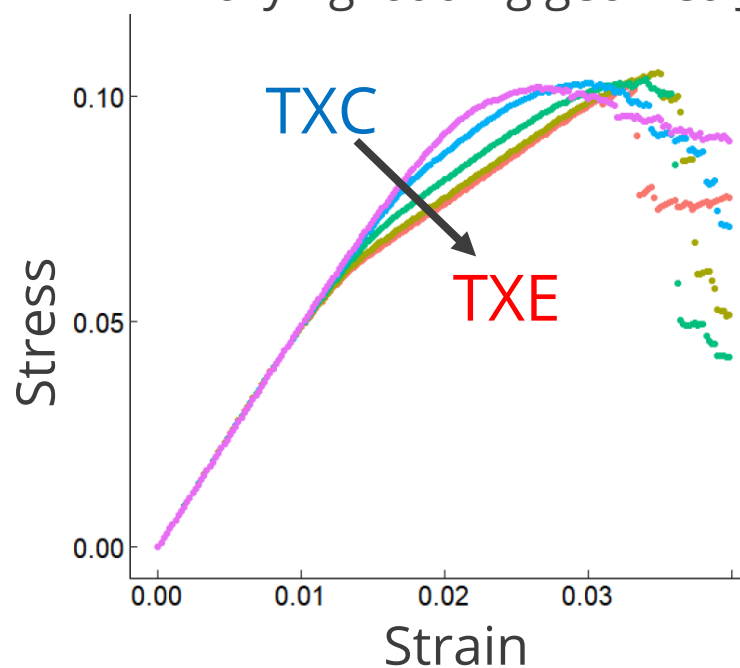


Extending framework to capture damage and crack percolation

Simulations allow us to identify yield while simultaneously tracking crack growth in binder & quantifying damage

Complete spatial-temporal history of damage provides a new perspective on complex mechanical problems

See emergence of damaged modulus by varying loading geometry



Final percolating crack in binder in pure shear

Conclusions & Ongoing Work



- Multifaceted effort to understand and model mechanical behavior:
 - ViscoPlastic-ViscoSCRAM macroscale constitutive (strength) model with 3 key features: Viscoelasticity + SCRAM damage + viscoplasticity
 - Plasticity is an additional inelastic mechanism that limits stress development in the material which reduces the rate of damage and also produces positive volume strain (dilatation)
 - Improved agreement with intermediate strain rate SHPB test compared to model without plasticity

- Open Research Questions:
 - How best to capture coupling between the various deformation mechanisms (cracking, porosity opening, etc) in the macroscale model?
 - Best practices for calibration?
 - Material failure criterion— $F(\text{damage})$, $F(\text{plasticity})$, $F(\text{both})$??
 - Mesoscale bonded particle models provide insight to deformation mechanisms, shape of yield surface, etc.



Thank You!

Extra Slides

➤ Solution Algorithm:

- (1) After converging at time t_n , input:
 Time and strain increments: $\Delta t, \Delta \epsilon$
 Stress state: $s_n, (\sigma_m)_n, s_n^{(k)}$
 Damage variable: c_n
 Functions of the damage variable: $\psi_n, \theta_n, \Lambda_n$
 History variable: ϵ_n^p
- (2) Viscoelastic predictor:
 Initialize plastic flow increment: $\Delta \lambda = 0$
 Compute: σ_{n+1}^{TR} from (37), (34) and (24)
 Compute: $\sigma_{n+1} = \sigma_{n+1}^{TR}$
 Compute: $\epsilon_{n+1}^p = \epsilon_n^p$
- (3) Yield surface check:
 IF $f(\sigma_{n+1}, \epsilon_{n+1}^p, D_n) > 0$, THEN GOTO (4)
 ELSE GOTO (5)
 ENDIF
- (4) Viscoplastic corrector:
 Solve for $\{\sigma_{n+1}, \epsilon_{n+1}^p\}$. See Box 2 for return map algorithm.
- (5) Damage variable, Maxwell stress, damage rate, and damage variable functions updates:
 Update: c_{n+1} from (27)
 Update: $s_{n+1}^{(k)}$ from (29)
 Update: \dot{c}_{n+1} from (7)
 Update: ψ_{n+1} from (18)
 Update: θ_{n+1} from (19)
 Update: Λ_{n+1} from (20)