

Effects of Hydrogen on Deformation Evaluated with EBSD of Single Crystal Austenitic Stainless Steel

Brian Kagay^{1,2}

Coleman Alleman¹, Brandon Talamini¹, Chris San Marchi¹

¹Sandia National Laboratories

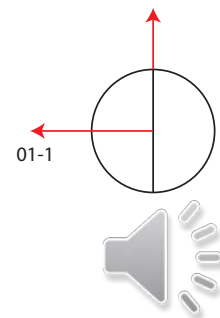
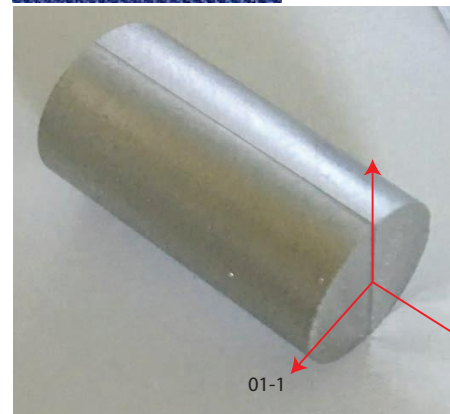
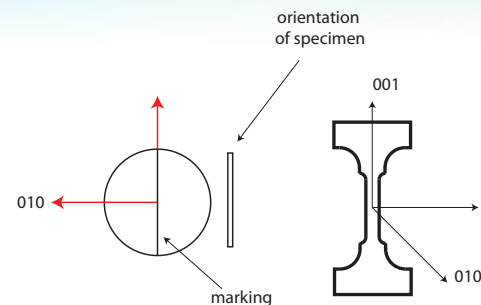
²Now at MPA University of Stuttgart

February 28th, 2022



Evaluated hydrogen effects on deformation mechanisms of 316L single crystals

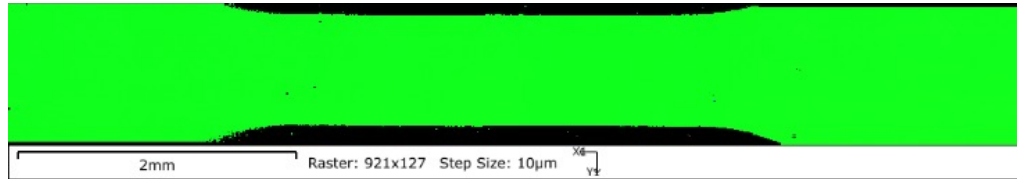
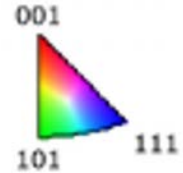
- Mini-dogbone tensile samples extracted from 316L single crystal cylinders with three different orientations
- Samples polished prior to testing to enable EBSD
- Half of the tensile samples pre-charged with hydrogen
- Strains measured with DIC and laser extensometer
- Interrupted testing to evaluate evolution of deformation
- Stress-strain curves and EBSD observations used to inform crystal plasticity modeling (yielding, hardening rate, grain rotation, etc.)



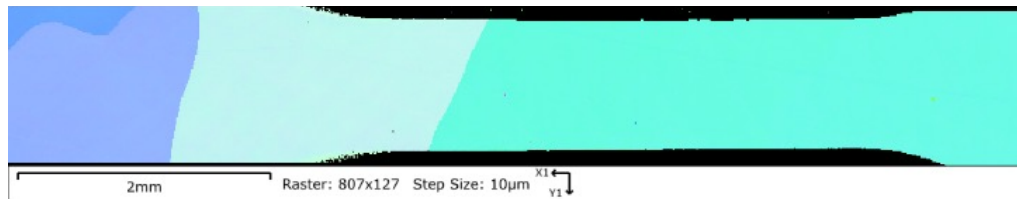
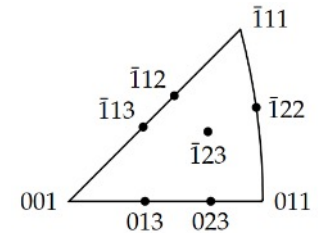
3 orientations parallel to tensile axis : $\langle 001 \rangle$, $\langle 011 \rangle$, $\langle 123 \rangle$



$\langle 001 \rangle$
4 initial active slip planes



$\langle 011 \rangle$
2 initial active slip planes

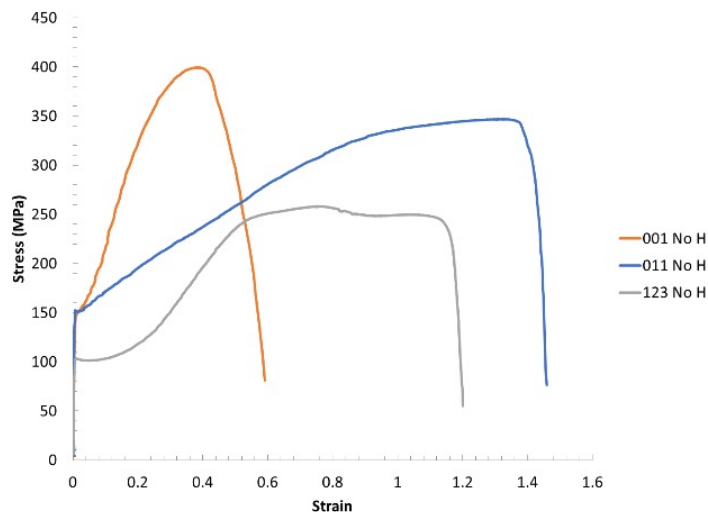


$\langle 123 \rangle$
1 initial active slip plane



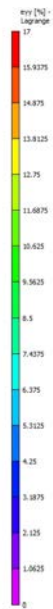
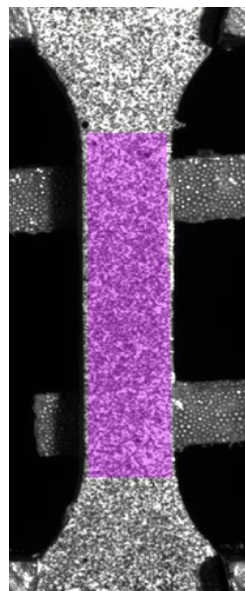
Differences in stress-strain behavior due to orientation and number of active slip planes

No H Tensile Curves

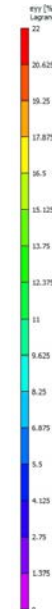
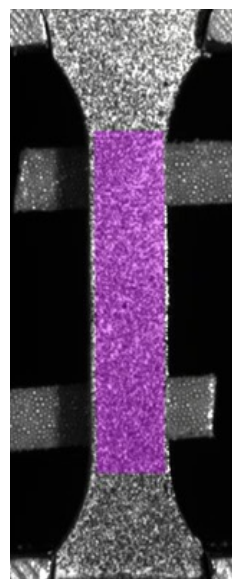


DIC videos for first 10% strain from interrupted tests

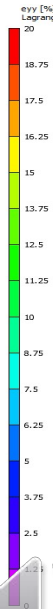
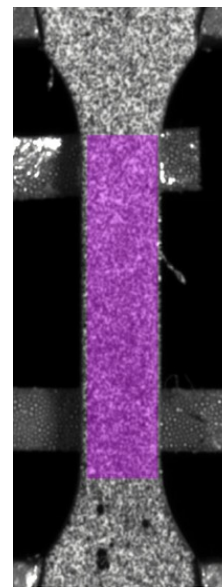
<001>



<011>

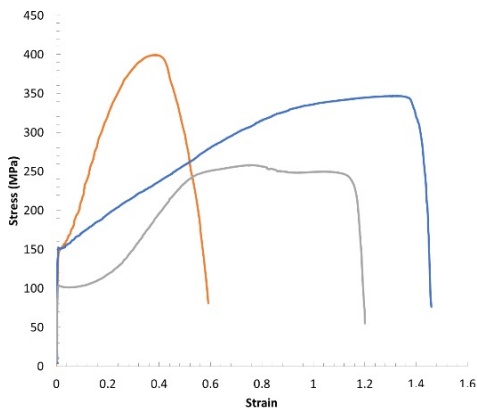


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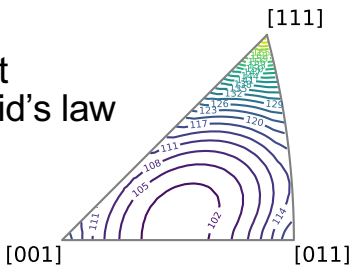
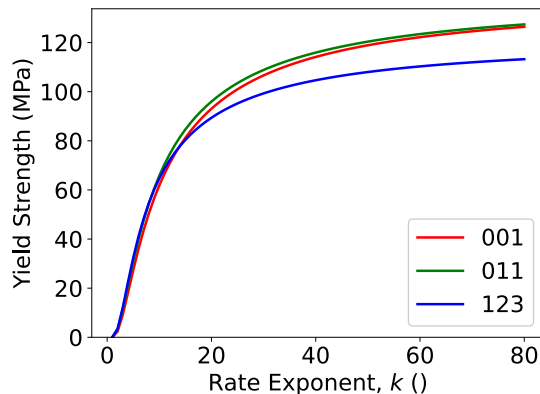
Yield stress, stages of hardening, hardening rate, and necking behavior are dependent on orientation and hydrogen content

No H Tensile Curves

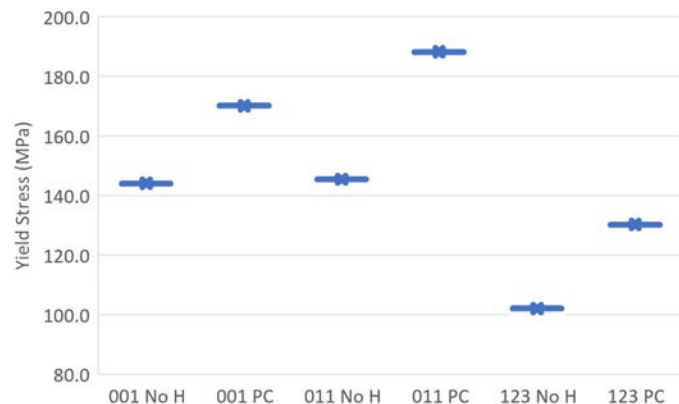


Yield stress does not correspond to Schmid's law

Yield behavior based on Schmid's Law

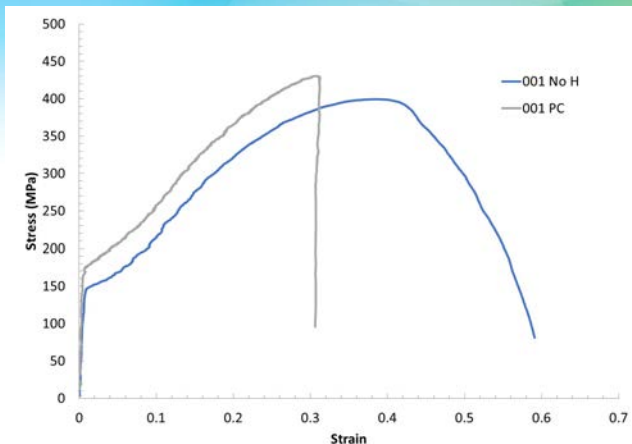


Amount of increase in yield stress due to hydrogen is dependent on orientation

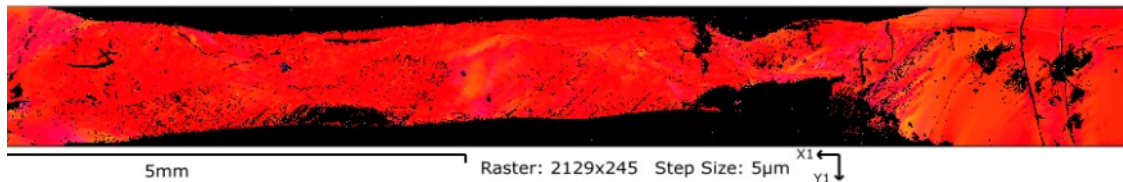


For $\langle 001 \rangle$ orientation hydrogen increased flow stress and decreased reduction of area

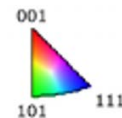
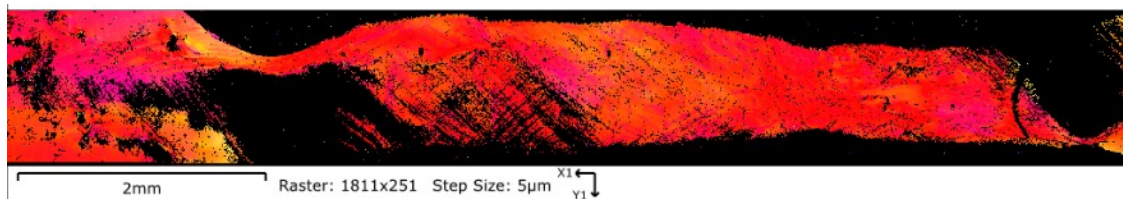
No significant twinning, martensite formation, or grain rotation observed through EBSD



$\langle 001 \rangle$ No H
Interrupted at
onset of necking

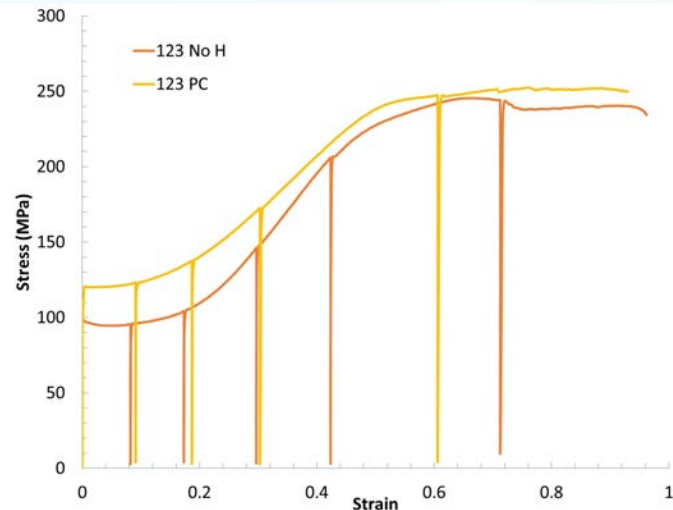
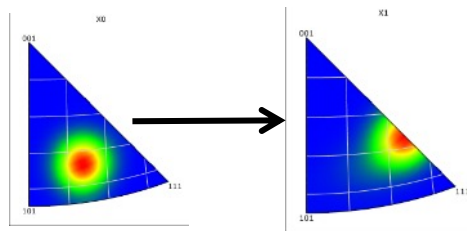


$\langle 001 \rangle$ PC
Interrupted at
onset of necking

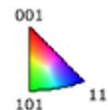
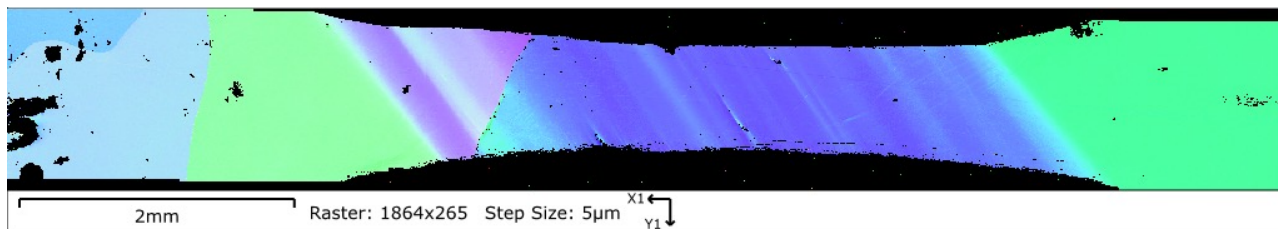
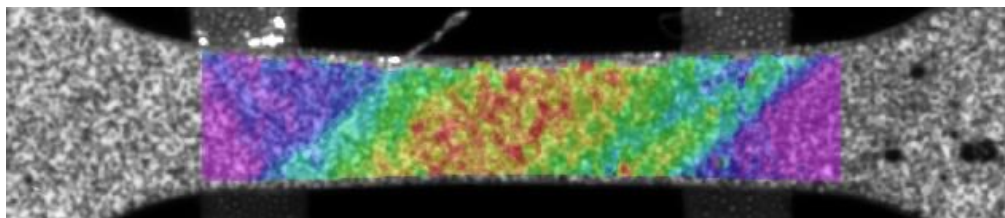


For $\langle 123 \rangle$ orientation hydrogen increased flow stress, decreased total hardening, and decreased load drop at onset of diffuse necking

- Grain rotated to $\langle 001 \rangle$ - $\langle 111 \rangle$ line and a second slip system activates
- No significant twinning or martensite formation observed through EBSD



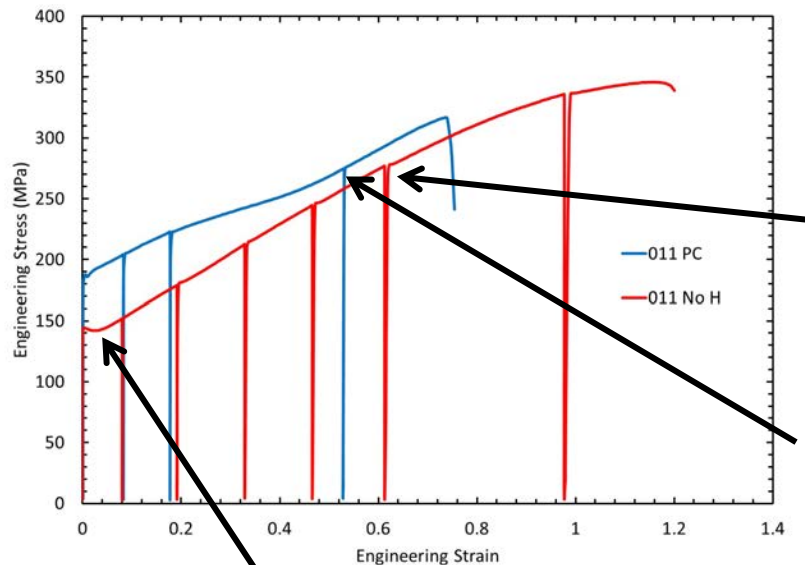
DIC and EBSD at 10% strain



Hydrogen affects deformation of $\langle 011 \rangle$ orientation prior to necking

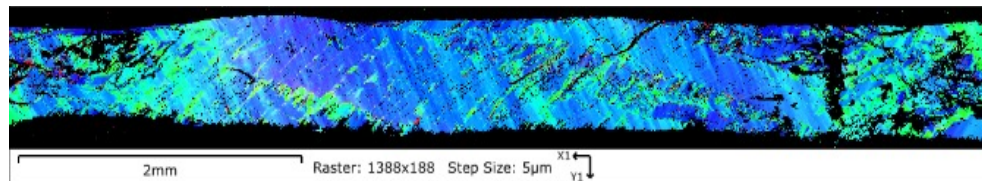
For $\langle 011 \rangle$ orientation, hydrogen

- decreased hardening rate between 20% and 50% strain
- caused twinning to occur at lower strains
- increased flow stress
- decreased uniform elongation
- decreased reduction of area

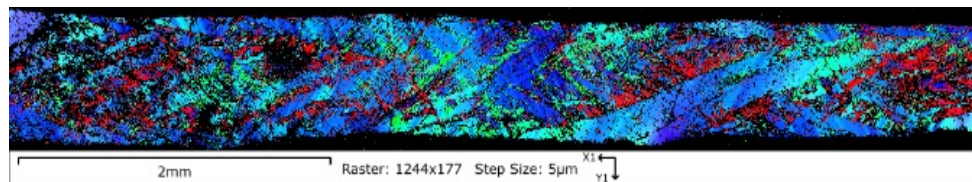


Small single slip stage I

$\langle 011 \rangle$ No H



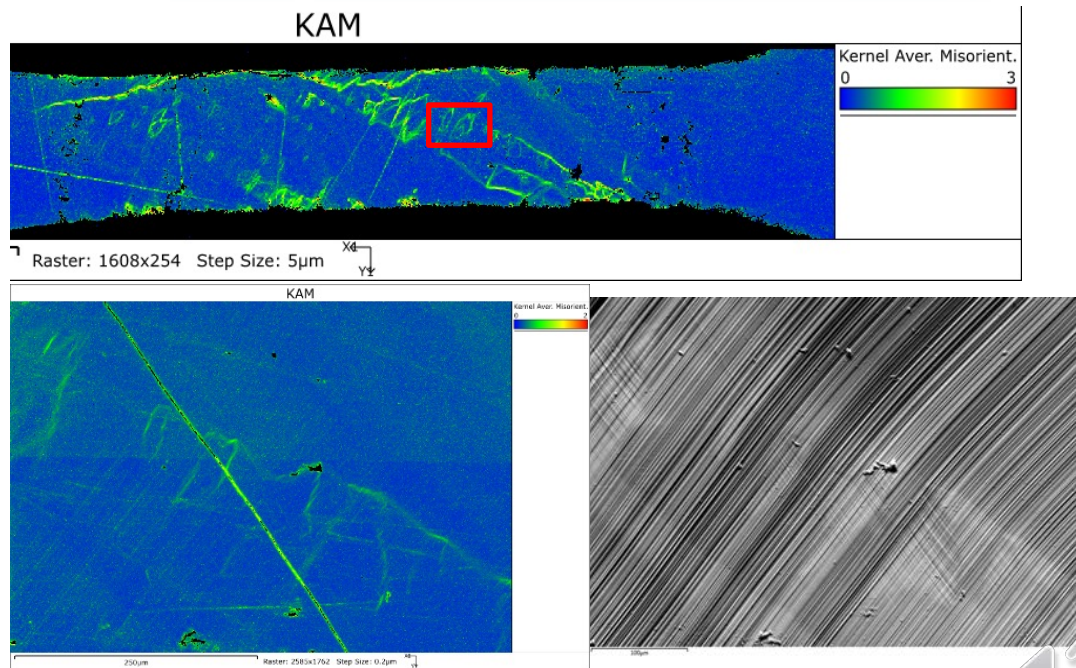
$\langle 011 \rangle$ PC



Significantly more twinning at 50% strain when hydrogen is present

For $\langle 011 \rangle$ at low strains the highest local misorientations (GNDs) are at edges of regions with slip lines in multiple directions

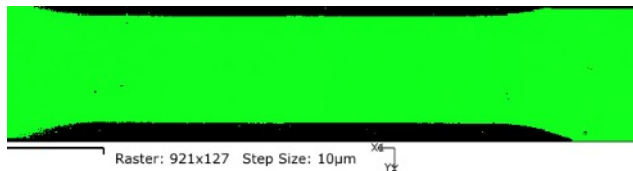
- Initially slip occurs primarily on one slip plane
- Shortly after yielding slip begins to occur on a second slip plane
- Dislocation interactions occur which can form obstacles to dislocation motion (Lomer-Cottrell locks)
- Geometrically necessary dislocations (GNDs) accumulate where slip is blocked at the obstacles



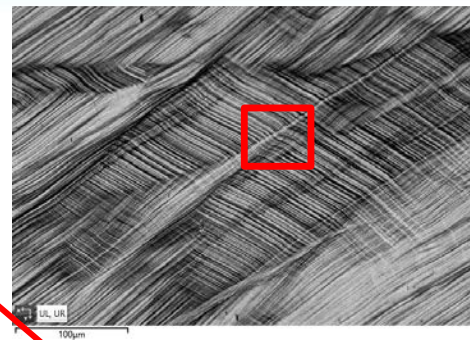
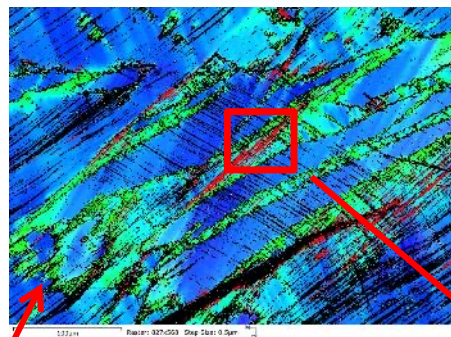
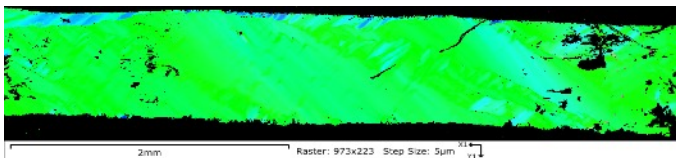
With further straining tensile axis of $\langle 011 \rangle$ rotates towards $\langle 111 \rangle$ direction

$\langle 011 \rangle$ No H

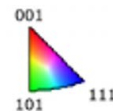
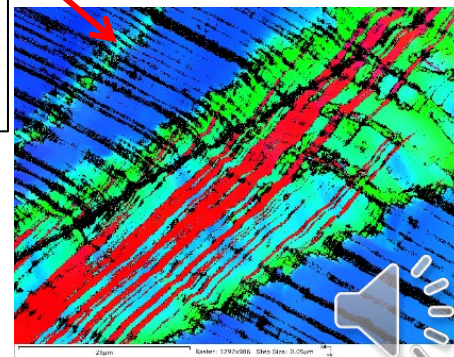
Pretest



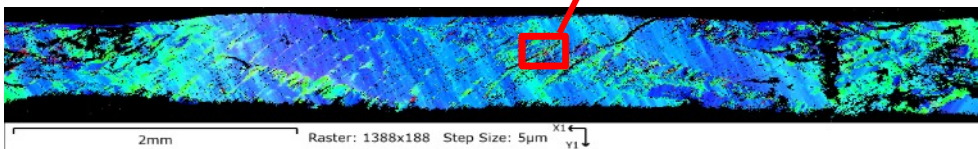
20% Strain
Some regions
rotating to
 $\langle 111 \rangle$



Twinning initially occurs at
low angle boundaries and
where rotation is not complete



60% strain
Most of crystal
rotated to $\langle 111 \rangle$
Twinning is visible

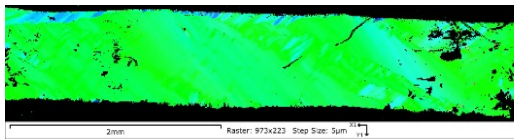


Hydrogen affects grain rotation and twinning planes for $\langle 011 \rangle$

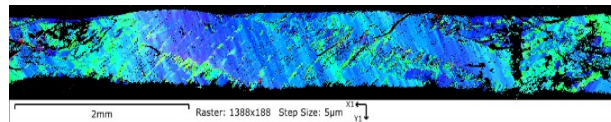
- For no H tensile axis rotates mostly towards a single $\langle 111 \rangle$, but $\langle 011 \rangle$ PC rotates towards two different $\langle 111 \rangle$.
 - These are rotations in the direction of the leading partial dislocations on the active slip plane
- Twinning occurs primarily on one plane with no H, but with H significant twinning occurs on two different planes

$\langle 011 \rangle$ No H rotates mostly to a single $\langle 111 \rangle$

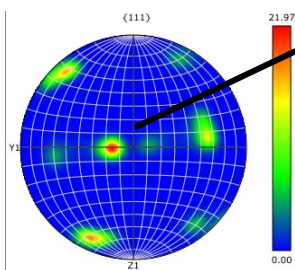
20% strain



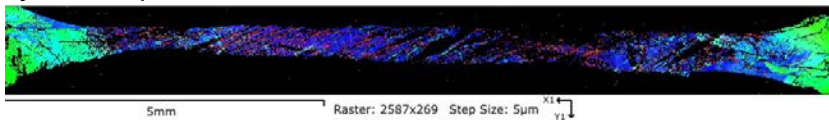
50% strain



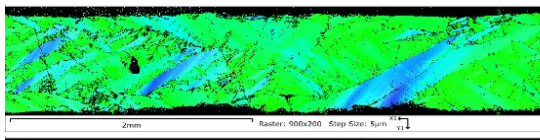
50% strain
 $\langle 111 \rangle$ pole figure
in tensile direction



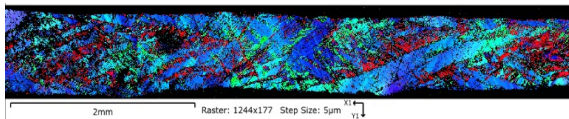
Necked. Twinning
mostly on one plane



$\langle 011 \rangle$ PC rotates towards two $\langle 111 \rangle$

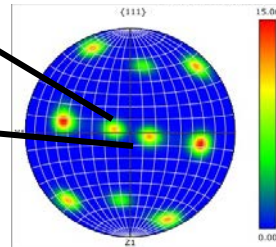


20% strain

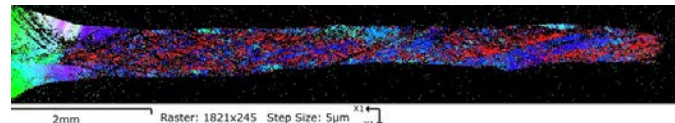


50% strain

50% strain
 $\langle 111 \rangle$ pole figure
in tensile direction

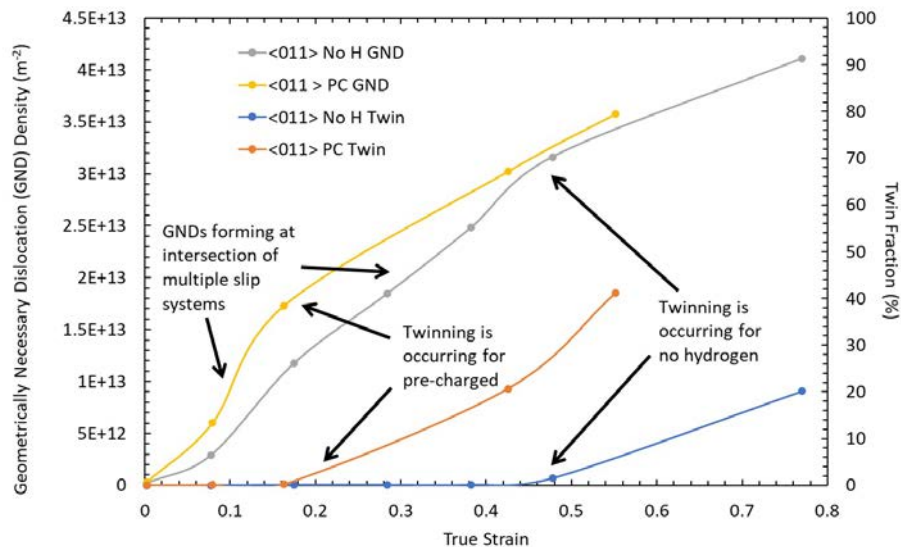


Failed. Twinning
on two planes



For $\langle 011 \rangle$ hydrogen causes twinning to occur at a lower GND density

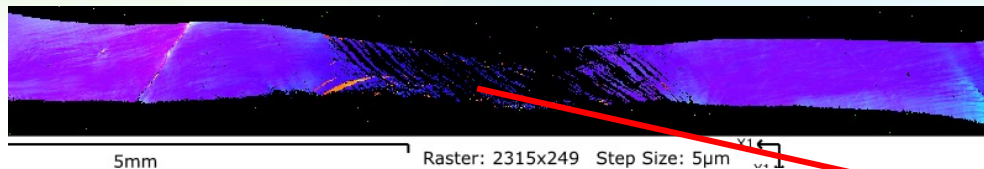
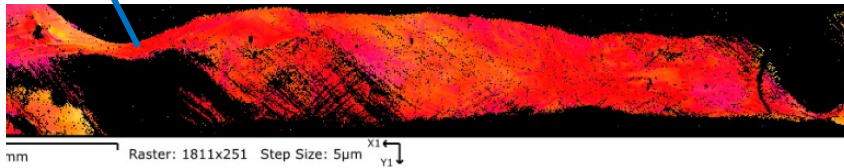
- GND density estimated from kernel average misorientation maps
- Twinning fraction is fraction of cells that are indexed as twinned orientation
- GND density increases more at lower strains with hydrogen
- Twinning commences at lower GND density and lower strain with hydrogen
- Hydrogen may cause obstacles to dislocation motion to form at lower strains which increases dislocation density and results in twinning due to blocking of movement of partial dislocations



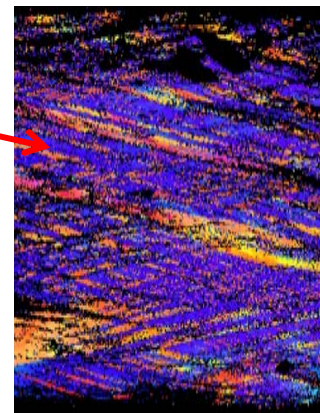
Twinning in necks of $\langle 011 \rangle$ and $\langle 123 \rangle$ specimens but not $\langle 001 \rangle$ No ϵ -martensite observed, even in neck of $\langle 011 \rangle$ PC specimen



$\langle 001 \rangle$ PC

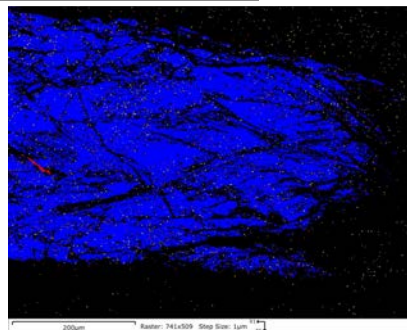
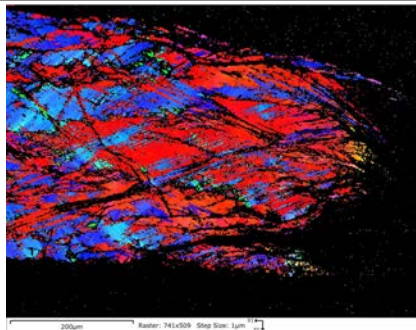


$\langle 123 \rangle$ PC

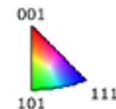


Two twinning directions in neck of $\langle 123 \rangle$ specimens

$\langle 011 \rangle$ PC
Small amount of α' -martensite in $\langle 011 \rangle$ specimen

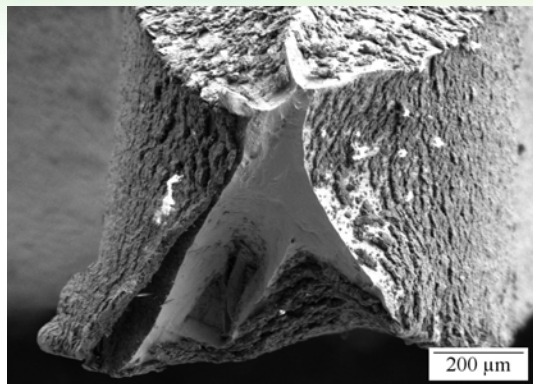


Phase Map
Blue = γ -austenite
Red = α' -martensite
Yellow = ϵ -martensite

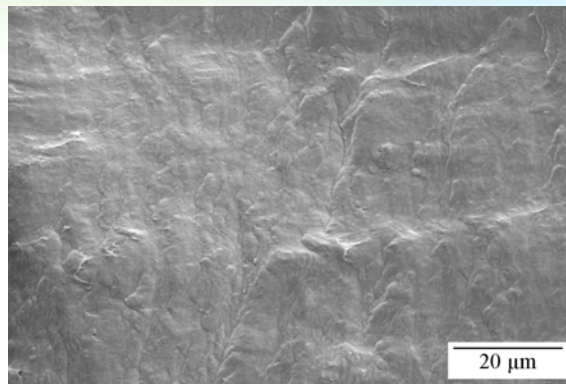


Fracture morphology dependent on orientation and presence of hydrogen

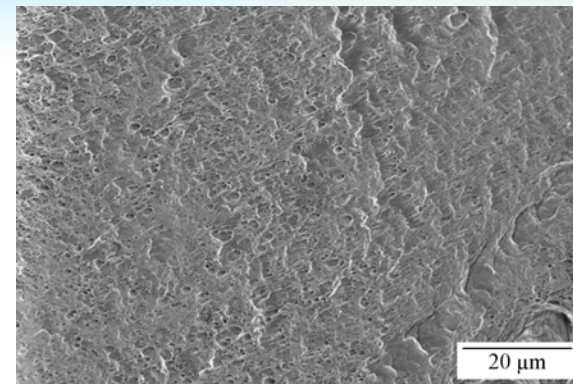
001 No H



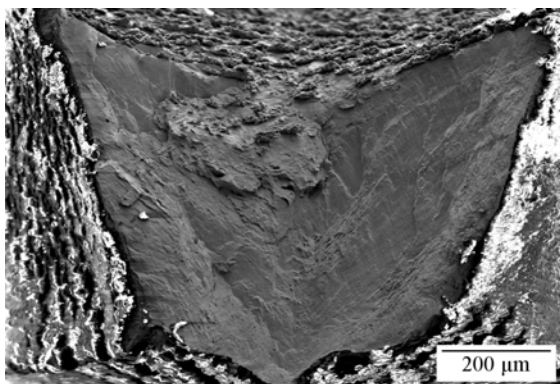
011 No H



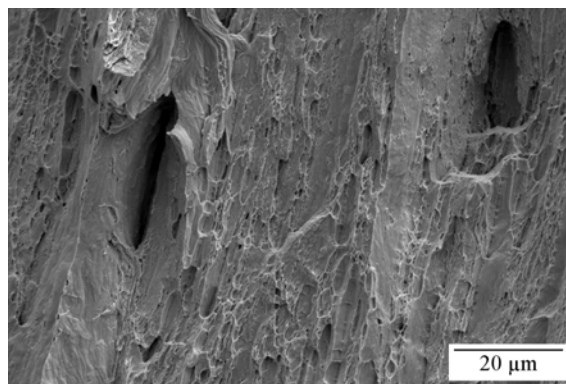
123 No H



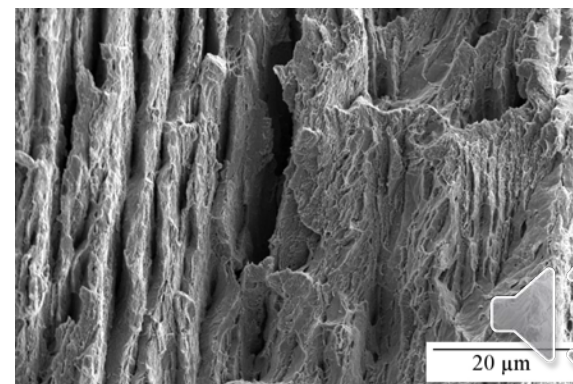
001 PC



011 PC



123 PC



Summary

- Tensile testing and EBSD of single crystal specimens with three different orientations parallel to the tensile axis ($\langle 001 \rangle$, $\langle 011 \rangle$, or $\langle 123 \rangle$) and with and without hydrogen was performed
- The yielding behavior of the three tested orientations did not correlate to Schmid's law and the increase in yield stress with hydrogen was dependent on orientation
- Hydrogen affected the flow stress and failure mechanisms of the $\langle 001 \rangle$ and $\langle 123 \rangle$ oriented single crystals, but did not significantly alter the deformation mechanisms prior to necking
- Hydrogen affected the stress-strain response and deformation mechanisms of the $\langle 011 \rangle$ specimen prior to necking
- For the $\langle 011 \rangle$ oriented specimen twinning occurred at lower strains and a lower GND density when hydrogen was present
- Twinning was observed in the necks of the $\langle 011 \rangle$ and $\langle 123 \rangle$ specimens, but not the $\langle 001 \rangle$ specimen, and no ϵ -martensite was observed
- The observed stress-strain behavior and deformation phenomena are being used to inform crystal plasticity modeling

Questions?

brian.kagay@MPA.uni-stuttgart.de

