



# Designing thin film microstructures using genetic algorithms

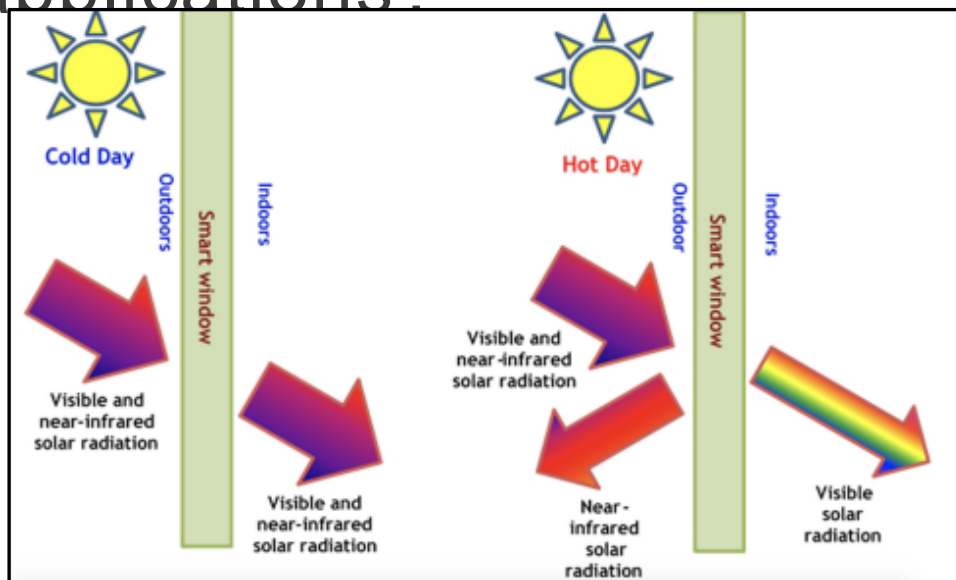
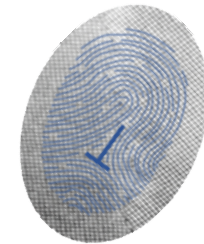
Saaketh Desai, Remi Dingreville

Center for Integrated Nanotechnologies

Sandia National Laboratories

TMS 2022 3rd March 2022

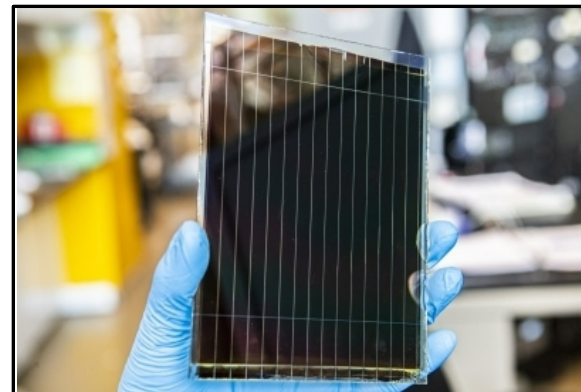
# How do we design thin films tailored for specific applications?



Source: [nist.gov](http://nist.gov)

Designing tailor-made thin films requires an understanding of processing-structure-property linkage

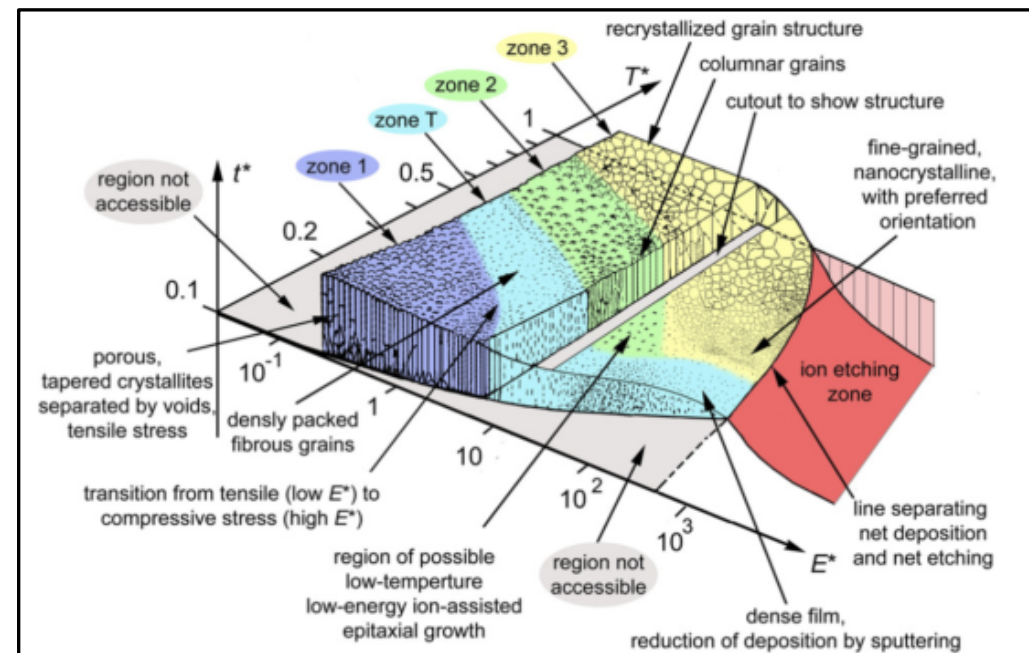
Structure zone diagrams relate processing conditions to microstructure



Source: [energy.gov](http://energy.gov)

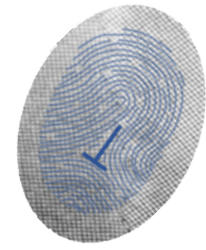


Source: [certechinc.com](http://certechinc.com)





# Microstructure formation in metallic alloy thin films



Binary metallic alloy microstructures governed by spinodal decomposition

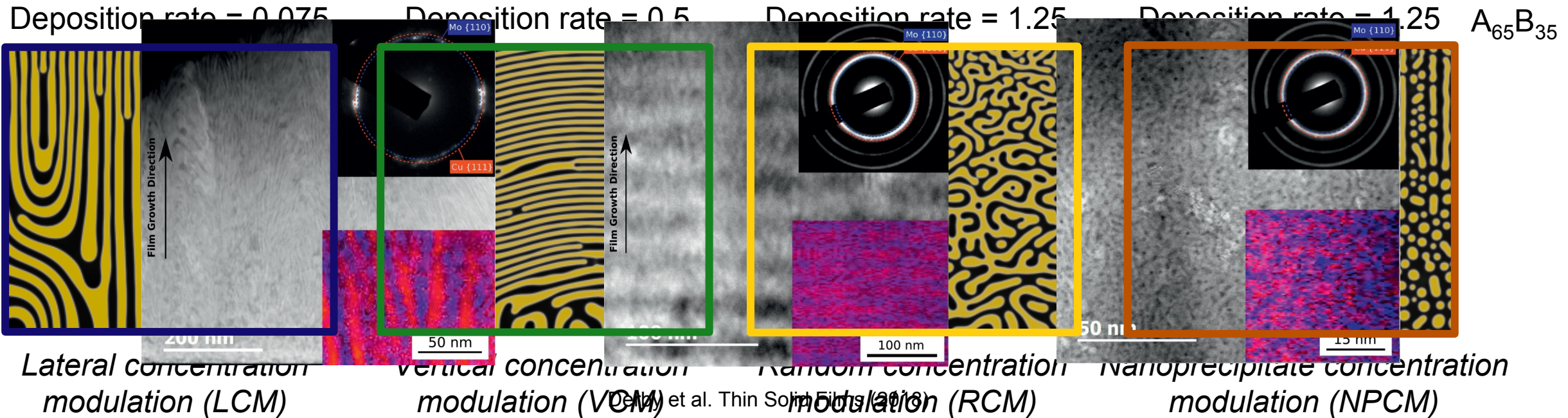
$$F = \int \left\{ f_\phi + \frac{\kappa_\phi}{2} (\nabla \phi)^2 + s(\phi) \left( f_c + \frac{\kappa_c}{2} (\nabla c)^2 \right) \right\} d\Omega \quad \frac{\partial c}{\partial t} = \nabla \cdot \left[ \mathbf{M}_c(\phi, c) \nabla \frac{\delta F}{\delta c} \right] \quad \frac{\partial \phi}{\partial t} = \nabla \cdot \left[ \mathbf{M}(\phi) \nabla \frac{\delta F}{\delta \phi} \right] + S(n(\phi))$$

Free energy of system

Stewart et al. *Acta Materialia* (2020)

$$\frac{\partial \rho}{\partial t} = \nabla \cdot [\mathbf{D}_\rho \nabla \rho] - \nabla \cdot [\rho \mathbf{v}] - S(n(\phi))$$

Evolution equations

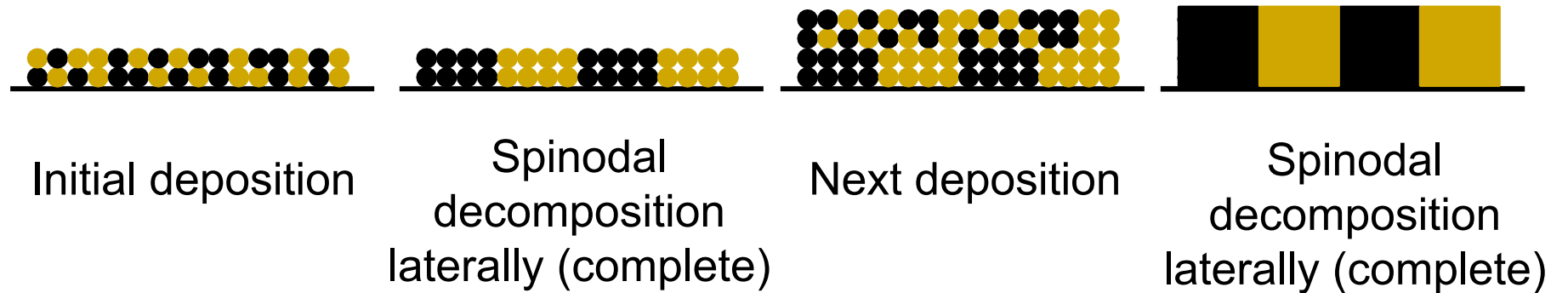
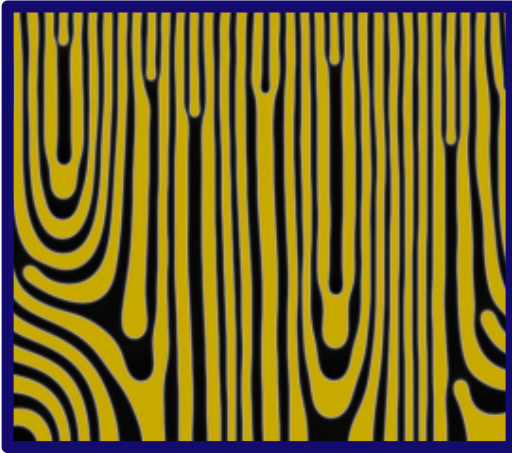


Phase field model simulates microstructure evolution for various deposition conditions

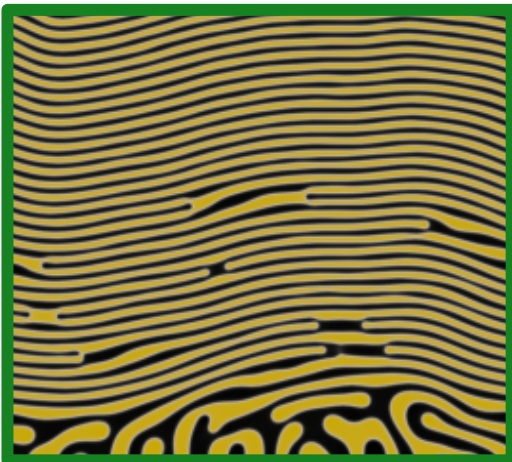
# Microstructures using constant deposition conditions



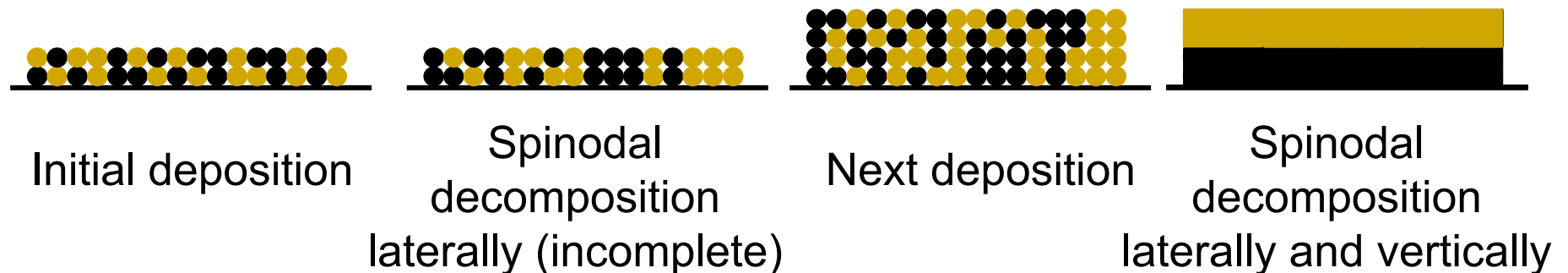
Deposition rate = 0.075



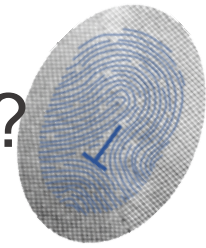
Deposition rate = 0.5



Competition between deposition/diffusion gives different microstructures

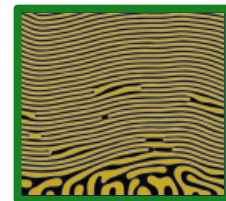
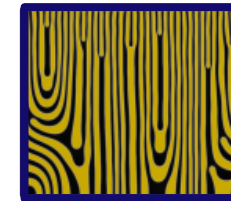
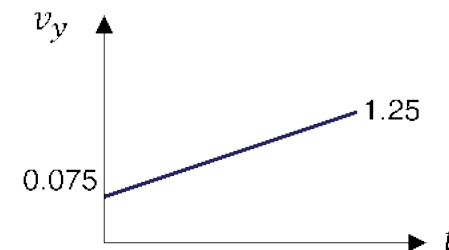
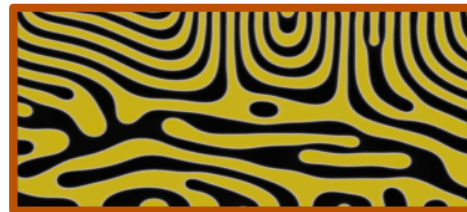
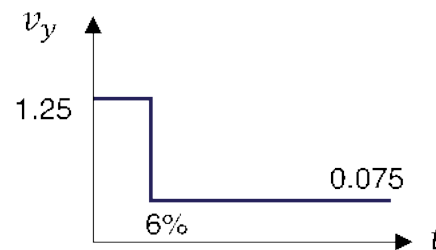
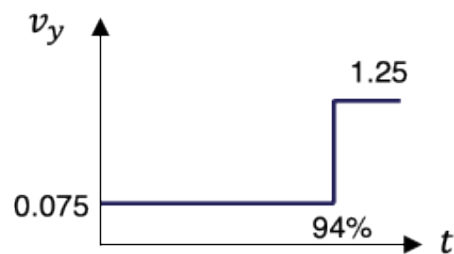


# How do we design metallic alloy thin film microstructures?

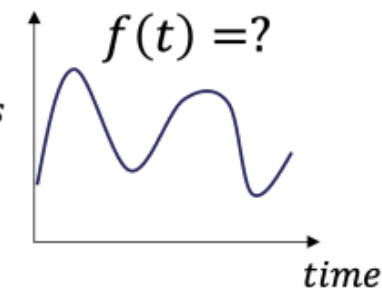


What processing conditions to use to obtain desired film microstructure?

Current SZDs only consider protocols that are constant in time



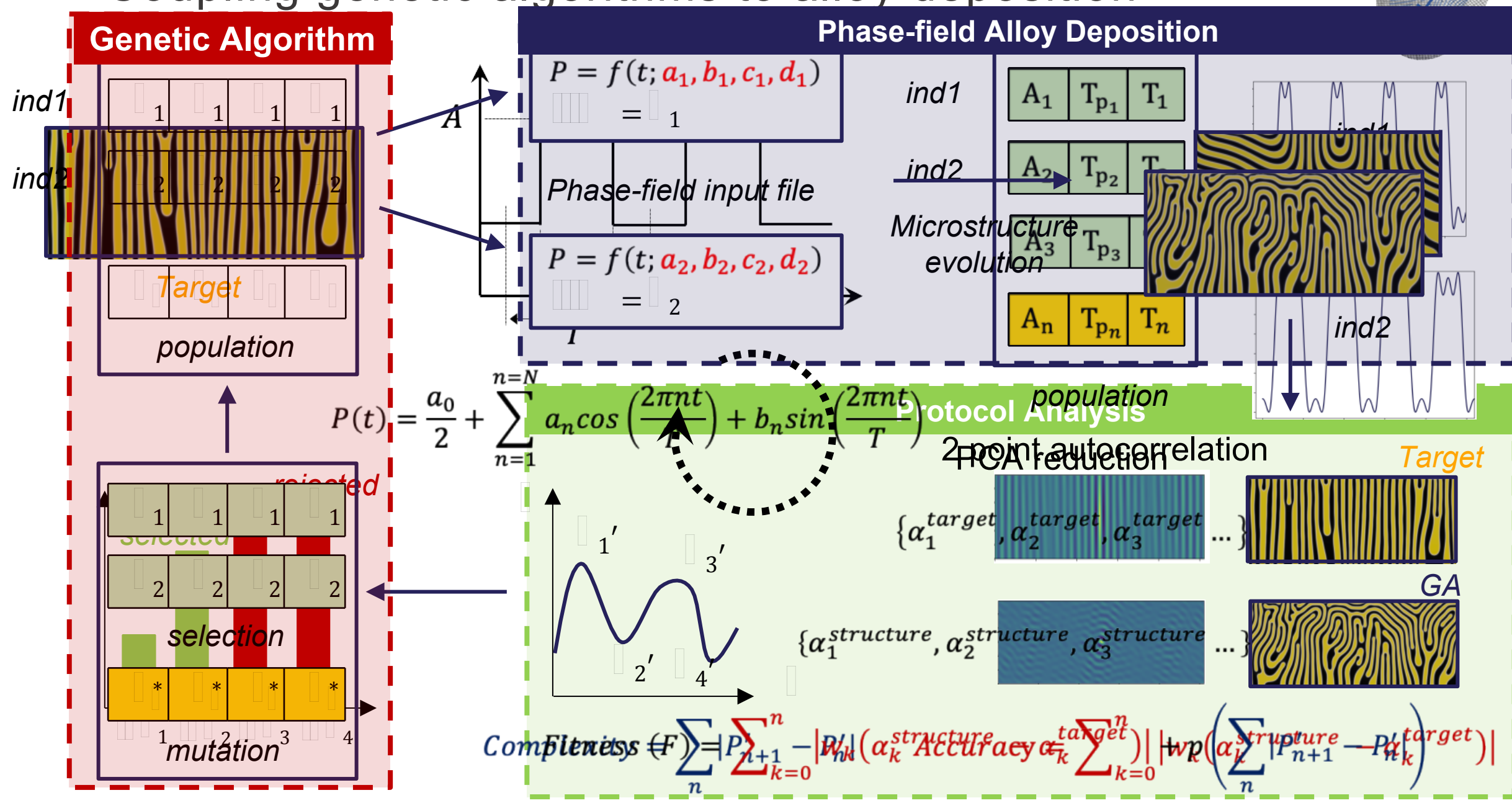
Deposition  
parameters



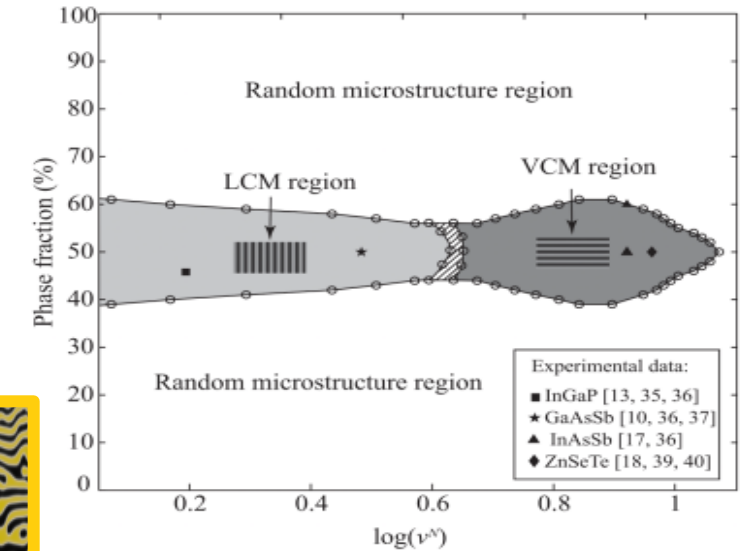
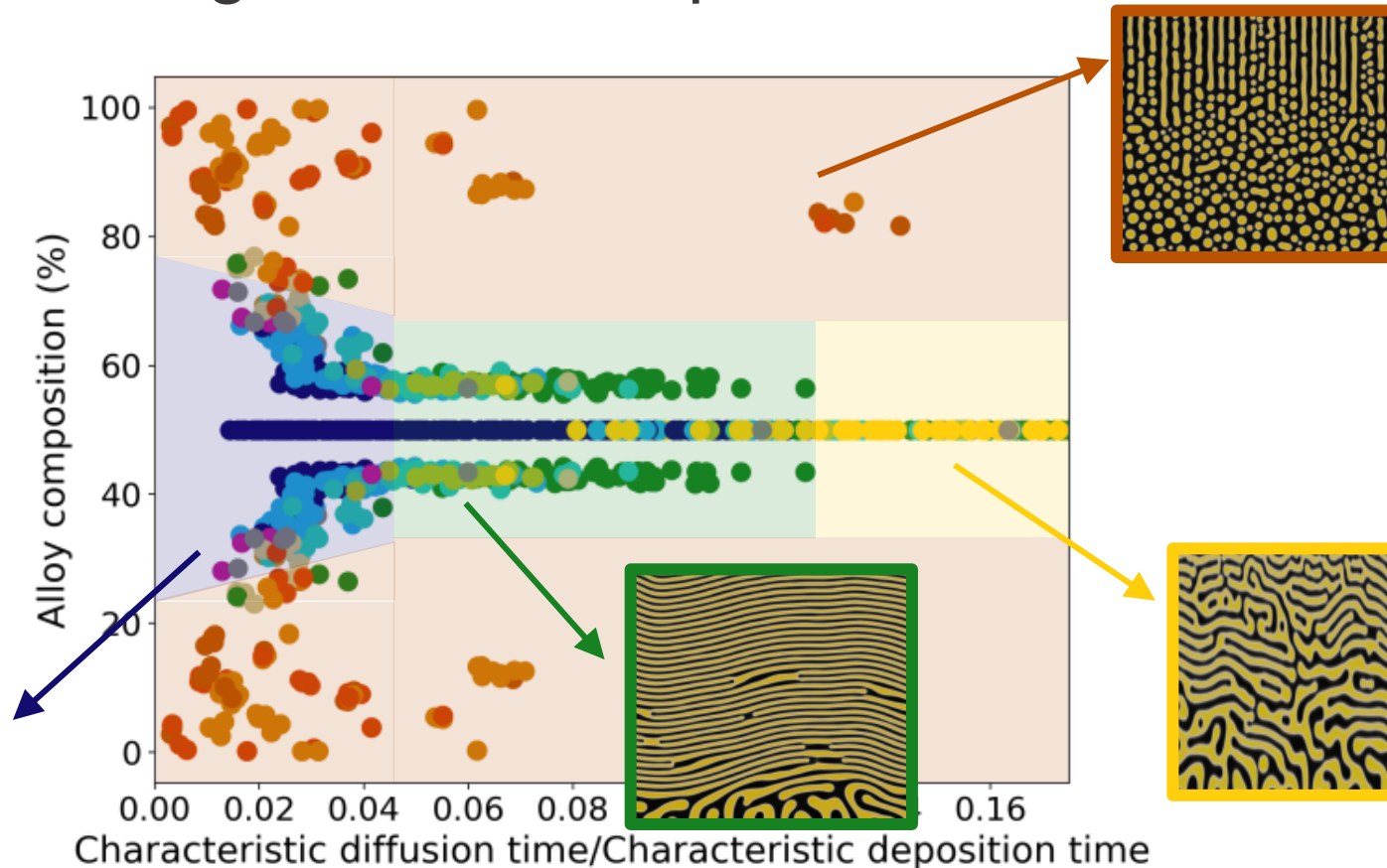
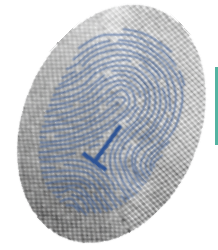
We use a genetic algorithm to discover time-dependent protocols that result in desired microstructure



## Coupling genetic algorithms to alloy deposition



# Re-creating constant deposition structure zone diagram



Lu, Yong et al. *Physical review letters* (2012)

- Low deposition/high diffusion rates lead to lateral concentration modulations
- High deposition/low diffusion rates lead to vertical concentration modulations
- Structure zone diagram agrees with previous phase field models and experiments

# Discovering time-dependent protocols

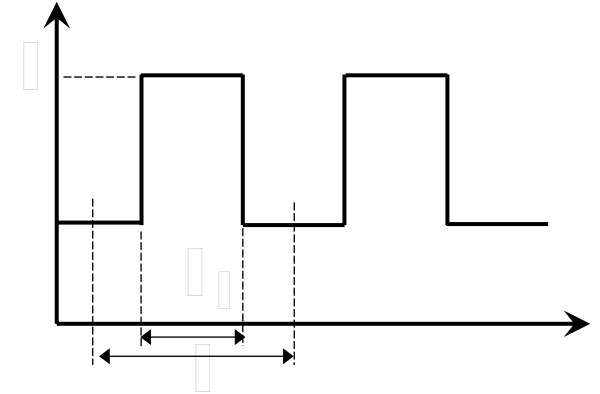


Search space: pulsed deposition rates

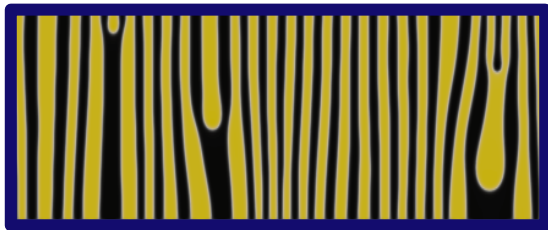
$$v_y^*(t) = \frac{a_0^*}{2} + \sum_{n=1}^{n=N} a_n^* \cos\left(\frac{2\pi n t}{T}\right) + b_n^* \sin\left(\frac{2\pi n t}{T}\right)$$

$$a_n^* = \frac{2A}{n\pi} \sin\left(\frac{n\pi T_p}{T}\right) \quad b_n^* = 0 \quad a_0^* = \frac{AT_p}{T}$$

Genetic algorithm parameters:  $A, T, f = \frac{T_p}{T}$



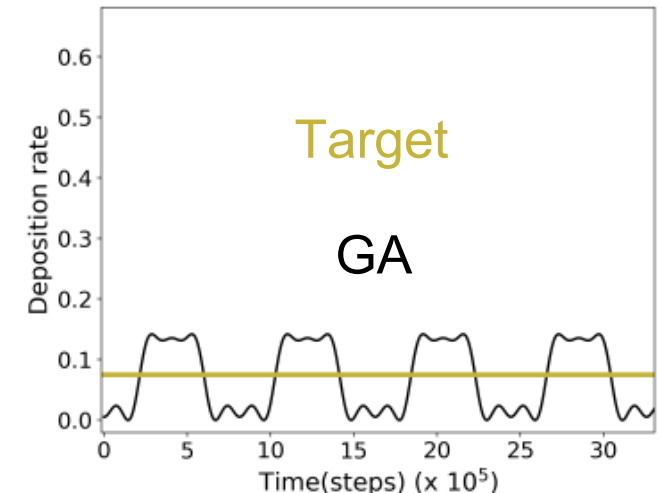
*Lateral concentration modulation (LCM)*



Target



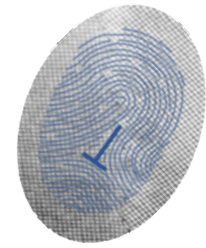
GA



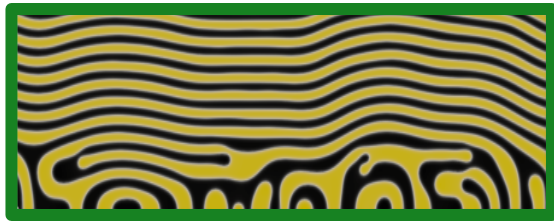
- Genetic algorithm discovers pulse protocol that results in target structure
- Deposition amplitude similar to deposition rate used to generate target



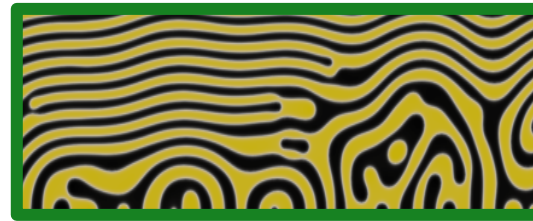
# Discovering time-dependent protocols



*Vertical concentration modulation (VCM)*



Target

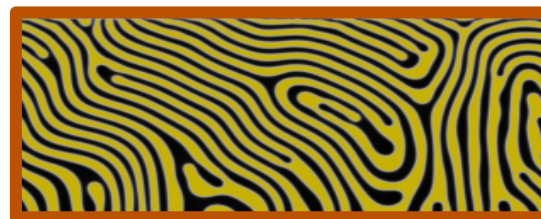


GA

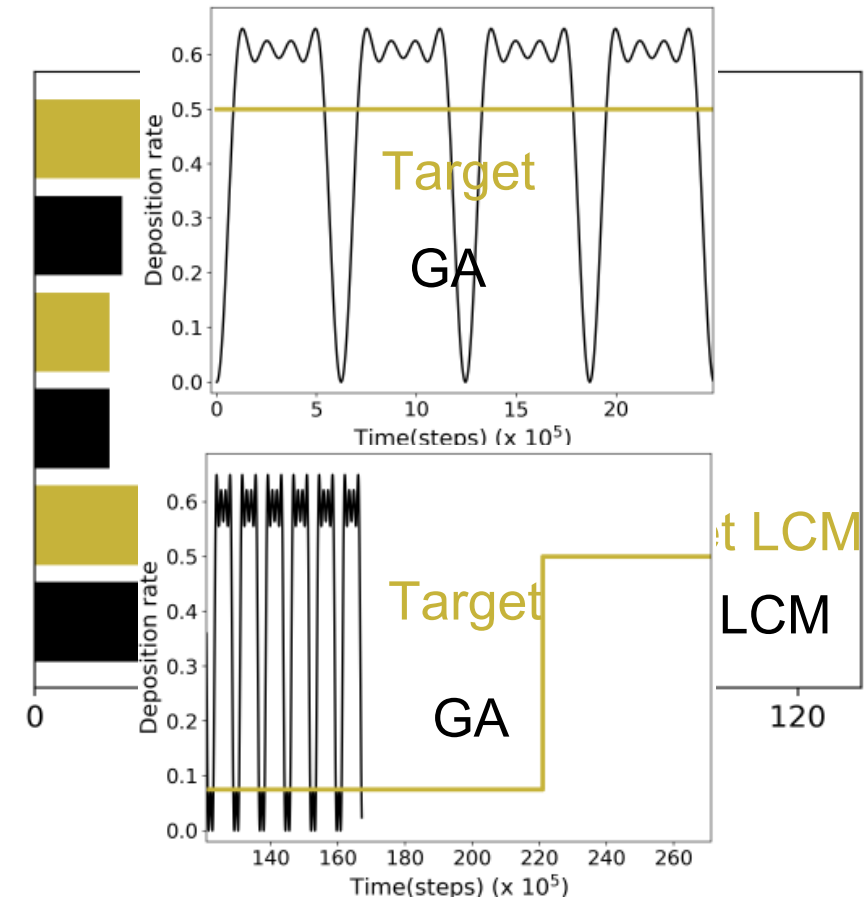
*Hierarchical microstructure*



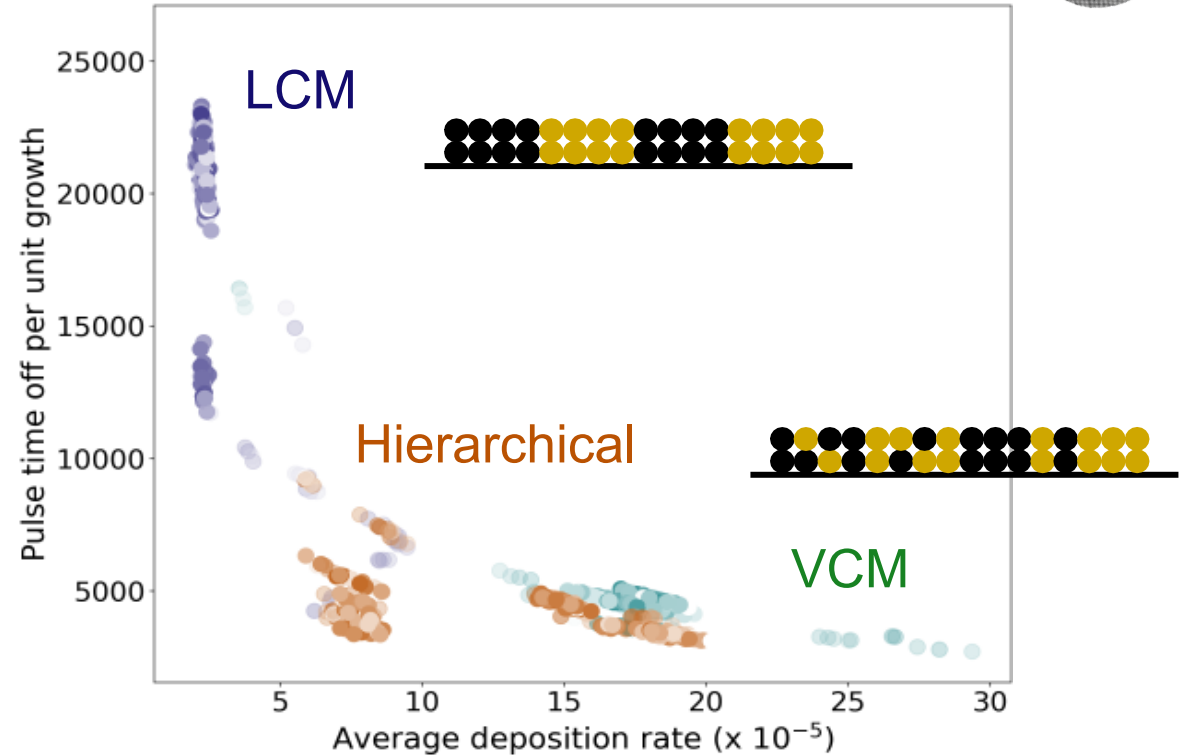
Target



GA



- Genetic algorithm discovers pulse protocol that results in target structure
- Time taken to achieve microstructure similar or lower than target protocol



- GA favors low amplitudes to generate LCM structures and high amplitudes for VCM structures
- Range of deposition rates can be used to get hierarchical structures
- Genetic algorithm learns deposition-diffusion trade offs

# Optimizing the deposition rate and the mobility



Search space: pulsed  
deposition rates and  
step function mobilities

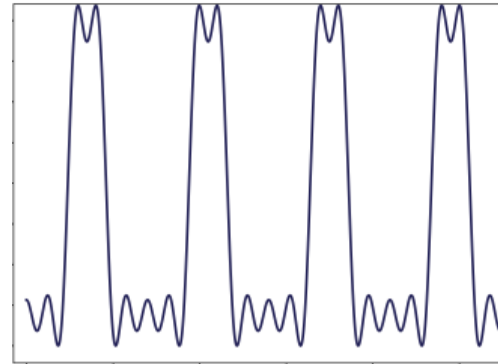
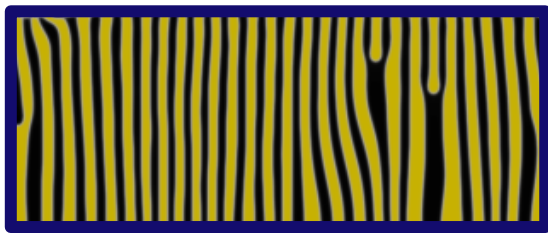
$$v_y^*(t) = \frac{a_0^*}{2} + \sum_{n=1}^{n=N} a_n^* \cos\left(\frac{2\pi n t}{T}\right) + b_n^* \sin\left(\frac{2\pi n t}{T}\right)$$

$$a_n^* = \frac{2A}{n\pi} \sin\left(\frac{n\pi T_p}{T}\right) \quad \text{with } \text{[symbol]}^* = 0 \quad a_0^* = \frac{AT_p}{T}$$

$$M(t) = \begin{cases} a & \text{if } t < t_0 \\ b & \text{if } t \geq t_0 \end{cases}$$

Genetic algorithm parameters:  $A, T, f = \frac{T_p}{T}, a, b, g = \frac{t_0}{T}$

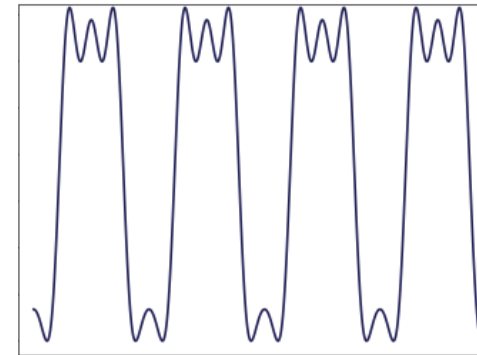
Target



Time

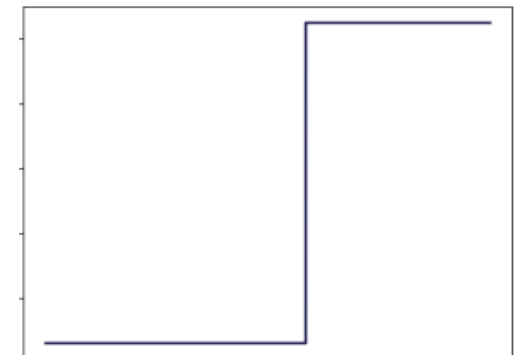
Deposition rate

Deposition rate



Time

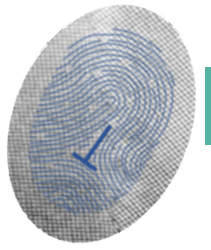
Mobility



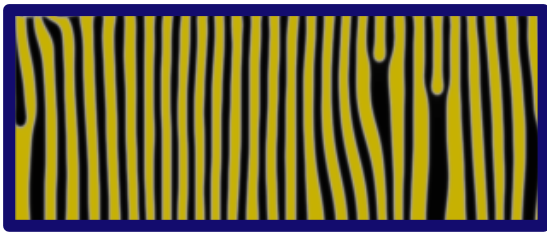
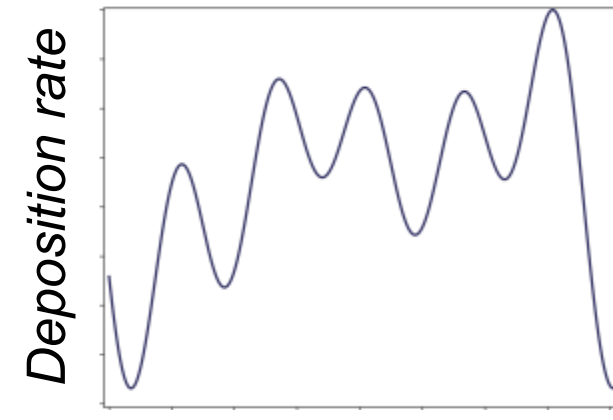
Time



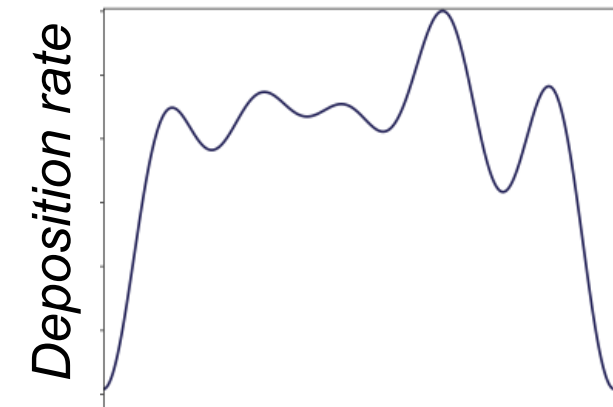
# Favoring simple protocols



$$\text{Fitness } (F) = \sum_{k=0}^n |w_k (\alpha_k^{\text{structure}} - \alpha_k^{\text{target}})| + p(\text{Complexity})$$

 $p_1$  $p_2$ 

*complex  
protocol*



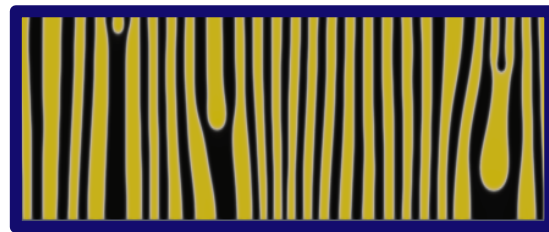
*simpler  
protocol*

Increasing weight on parsimony term  
gives near identical microstructure  
with a simpler protocol

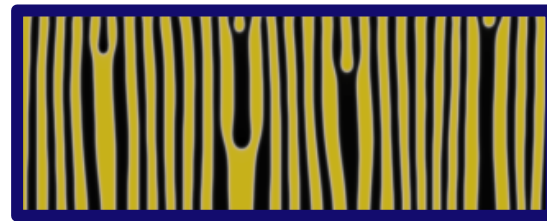
*Time (one period)*

# Take home message

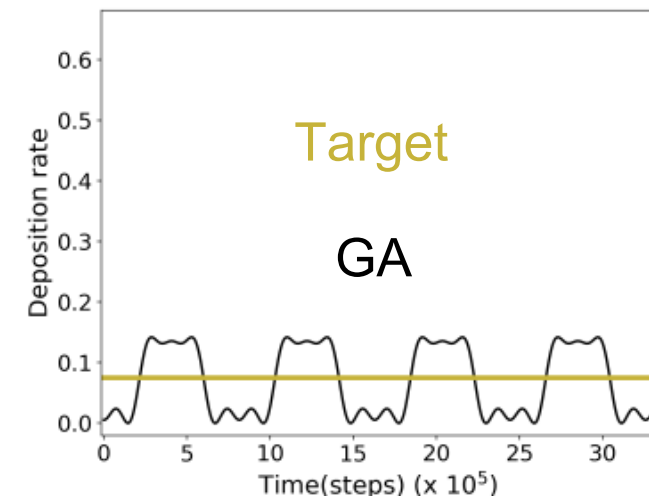
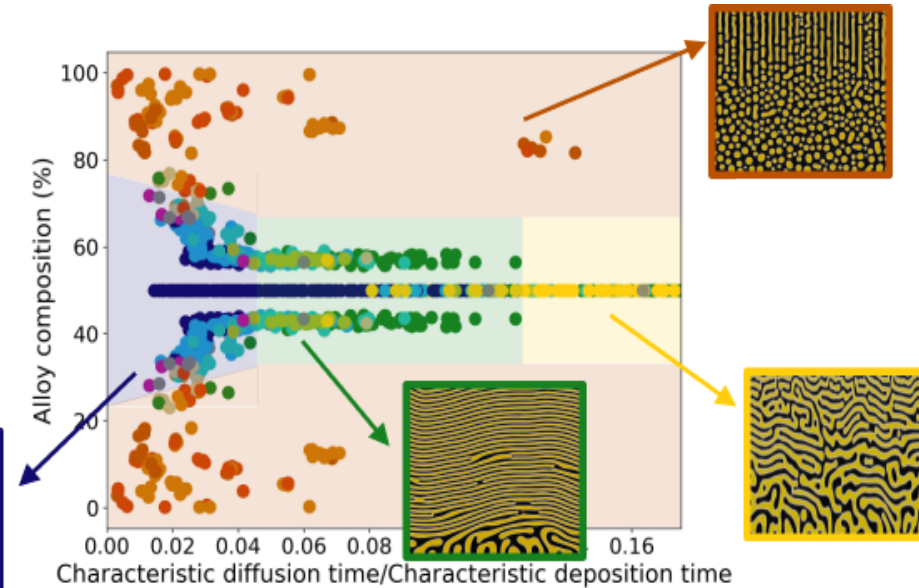
- Coupled genetic algorithm to phase field simulations to achieve targeted thin film microstructures
- GA achieves target microstructures for constant deposition scenarios
- GA discovers time-dependent deposition rates to achieve target microstructures, learning deposition-diffusion tradeoffs

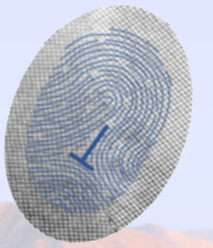


Target



GA





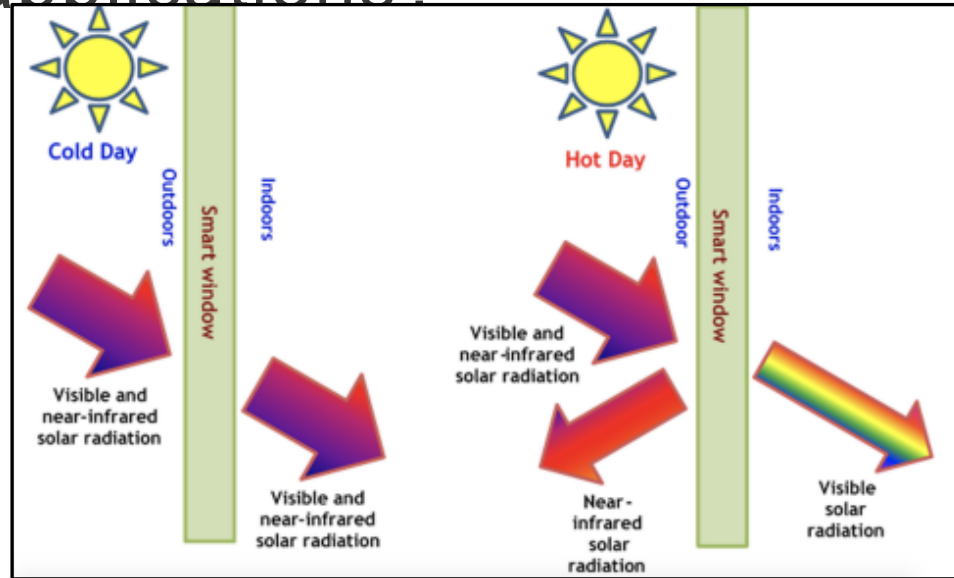
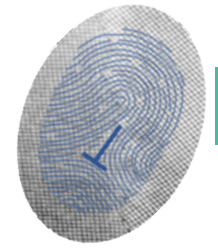
# Backup slides



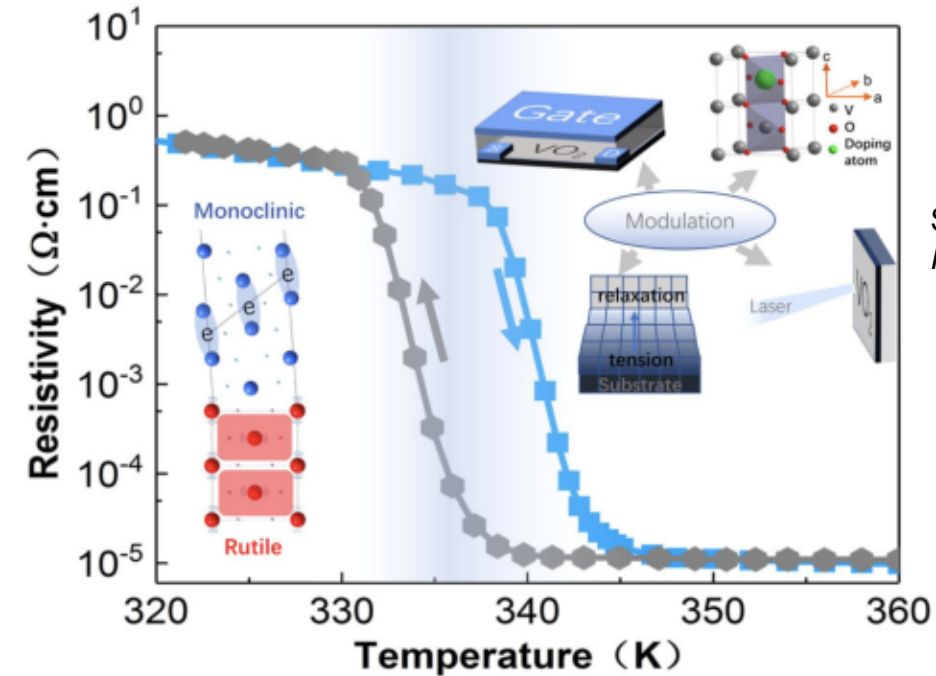
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Center for Integrated Nanotechnologies  
Sandia National Laboratories



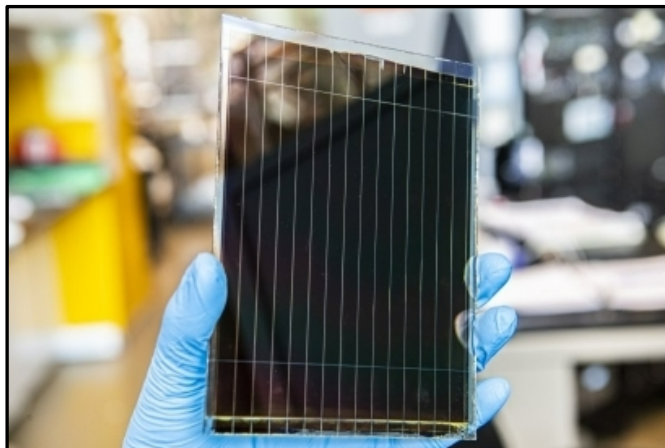
# How do we design thin films tailored for specific applications?



Source: [nist.gov](http://nist.gov)



Shao et al. *NPG Asia Materials* (2018)



Source: [energy.gov](http://energy.gov)



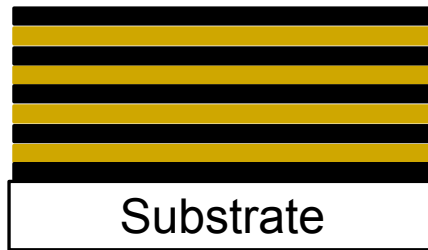
Source: [certechinc.com](http://certechinc.com)

Designing tailor-made thin films requires an understanding of processing-structure-property linkage

# Microstructure formation in metallic alloy thin films

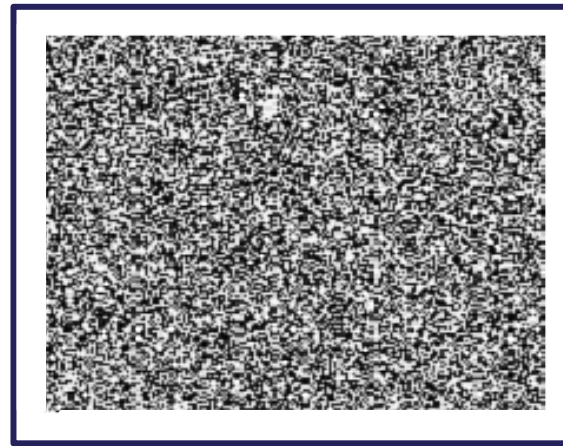
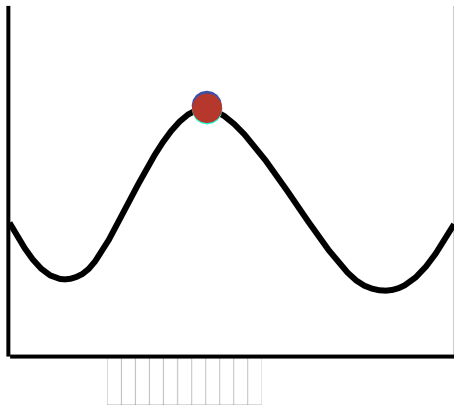


What processing conditions to use to obtain desired film microstructure?



Layer by layer deposition

Self-assembly via spinodal decomposition



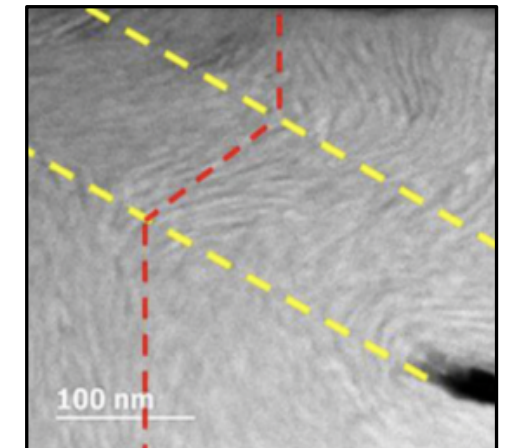
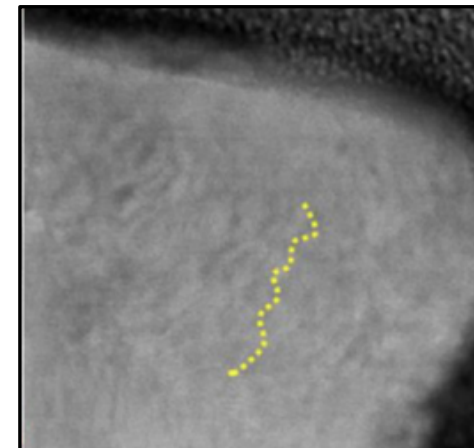
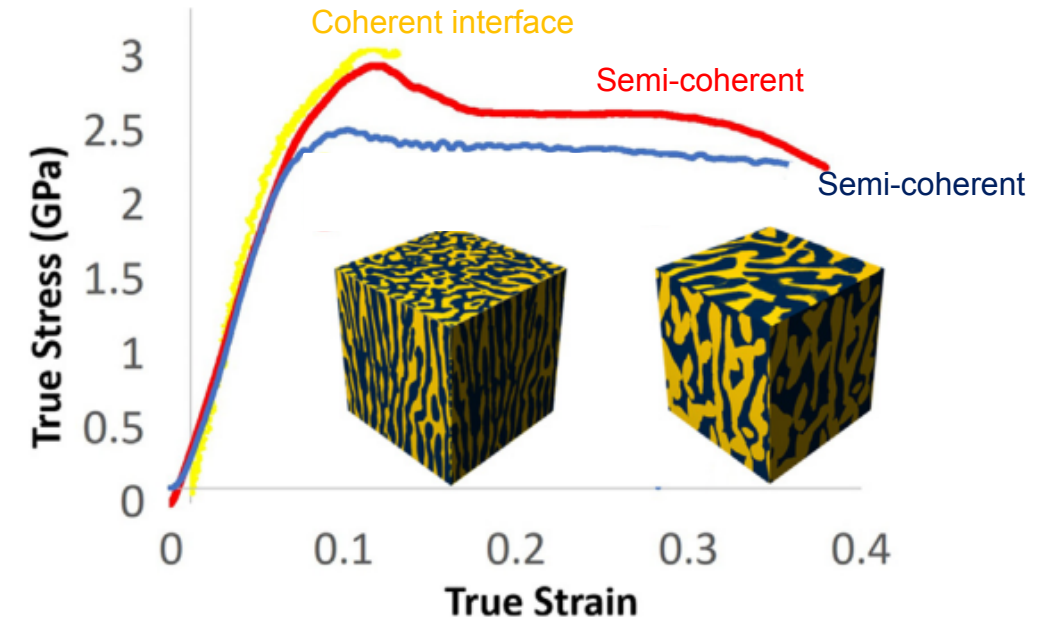
Source: [wikipedia](https://en.wikipedia.org/wiki/Spinodal_decomposition)

PVD experiments

Phase field simulations

Spinodal decomposition results in spontaneous concentration modulations

# Structure-property-processing relationships in thin films



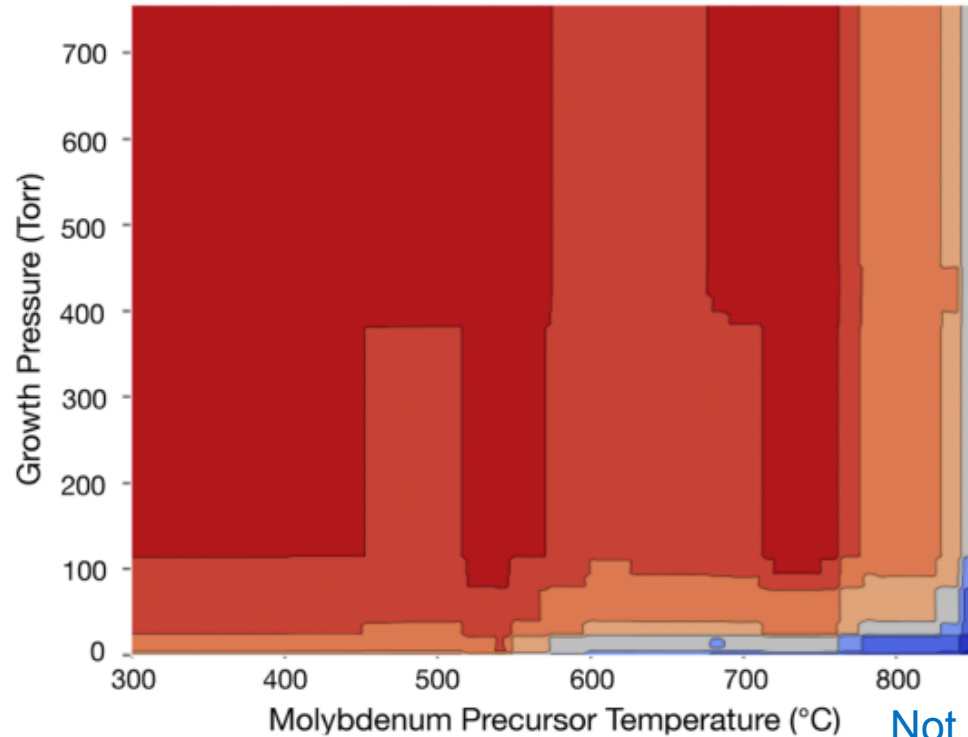


# High dimensional structure zone diagrams



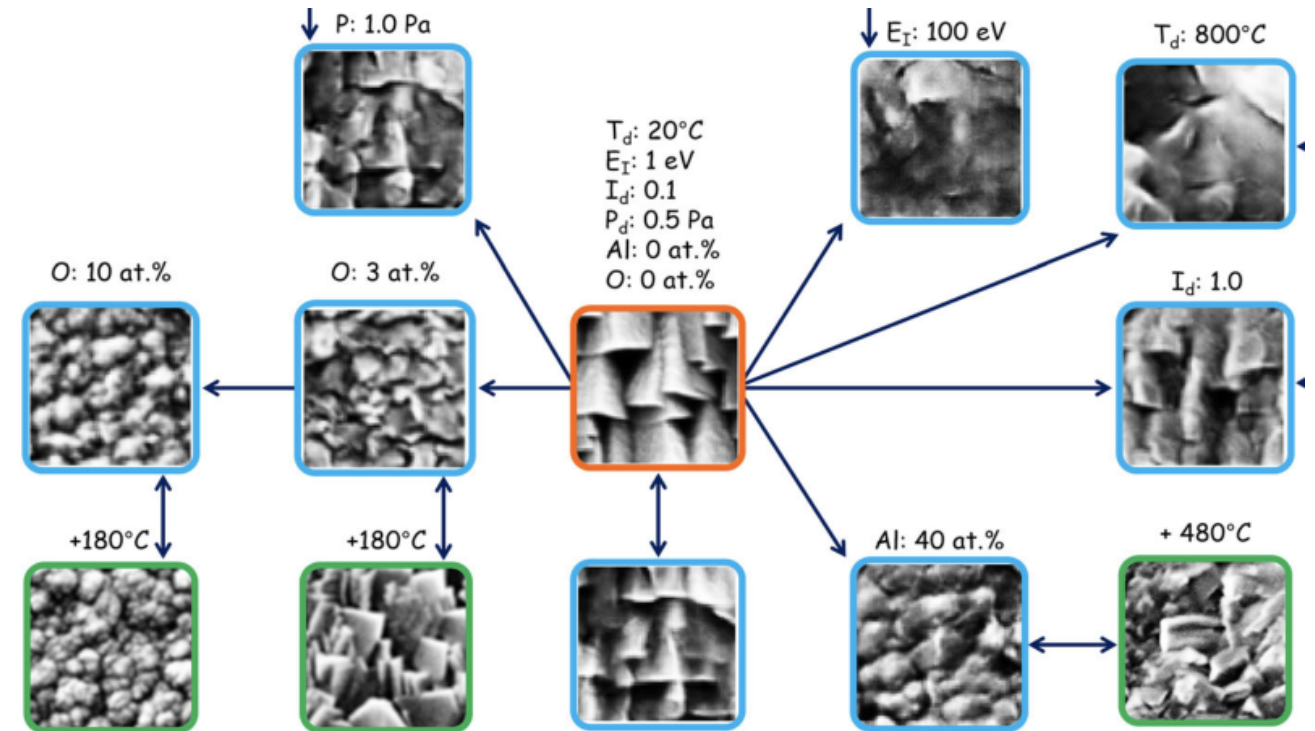
S precursor temperature = 145 °C  
Highest growth temperature = 730 °C  
Growth time = 52 minutes

Monolayer



Costine et al. *Journal of Applied Physics* (2020)

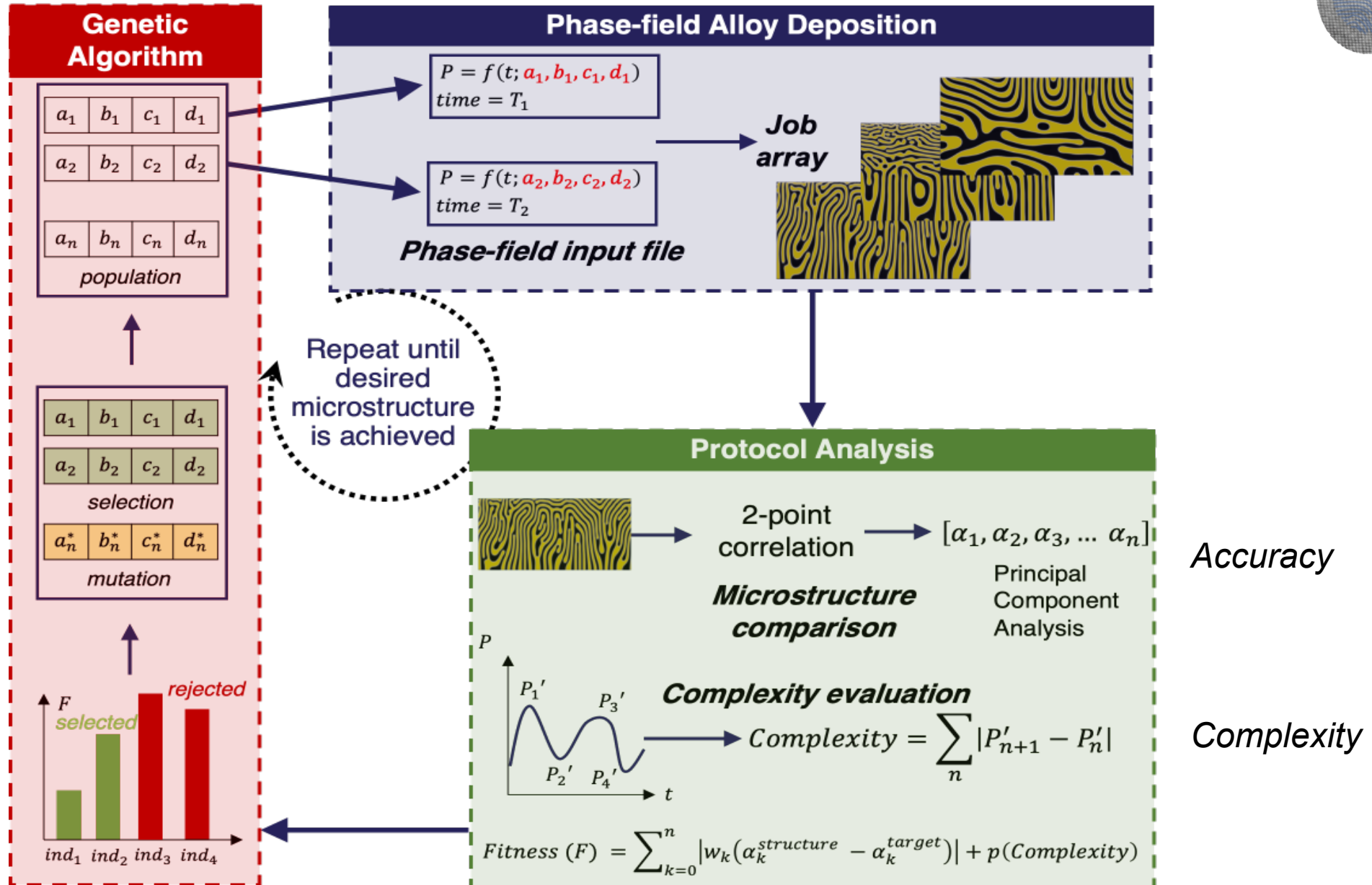
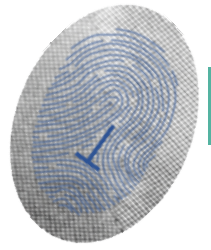
## Generative Adversarial Network based SZD



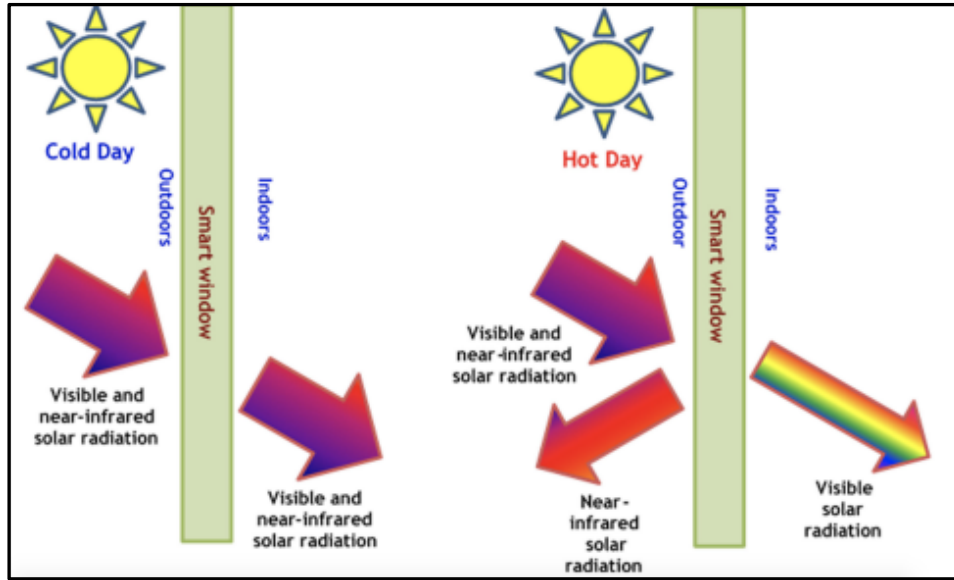
Banko et al. *Communications Materials* (2020)

ML methods can give high dimensional SZDs

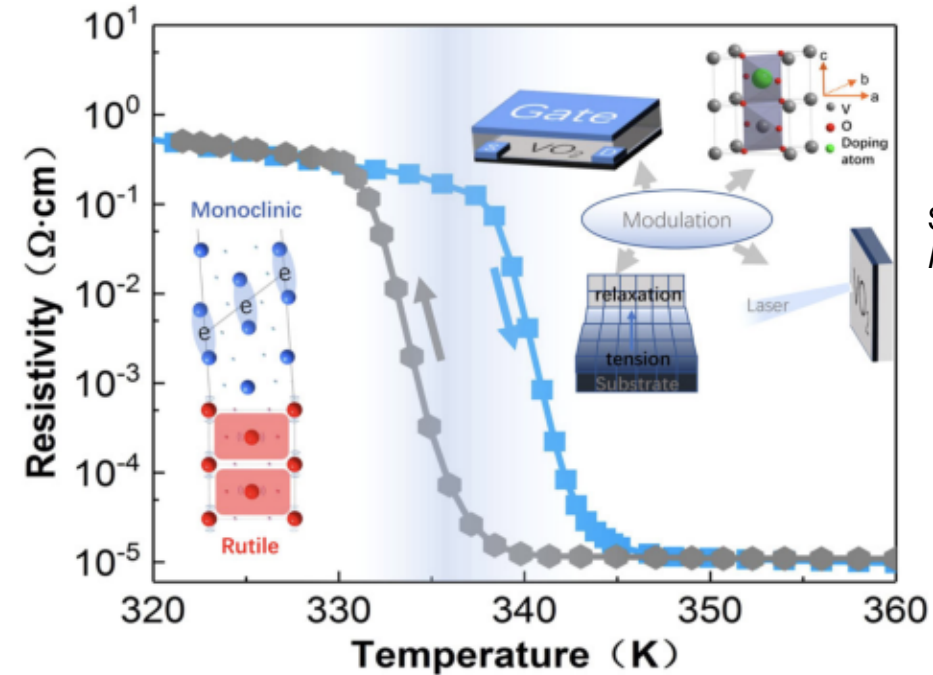
# Workflow



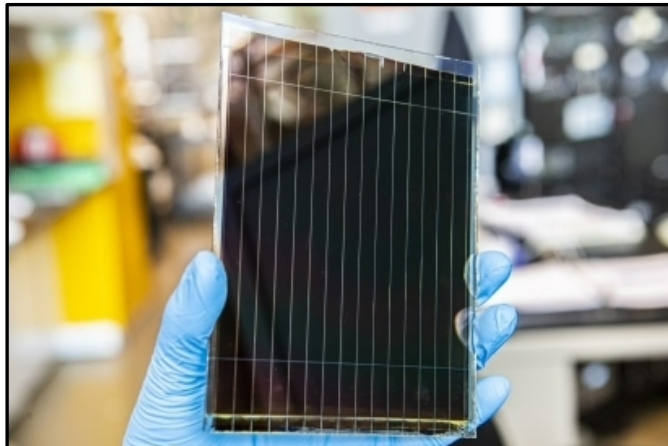
# Structure-property-processing relationships in thin films



Source: [nist.gov](http://nist.gov)



Shao et al. *NPG Asia Materials* (2018)



Source: [energy.gov](http://energy.gov)

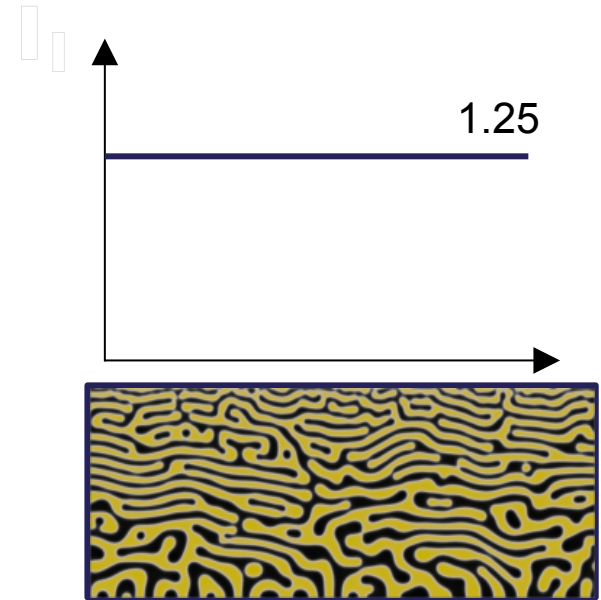
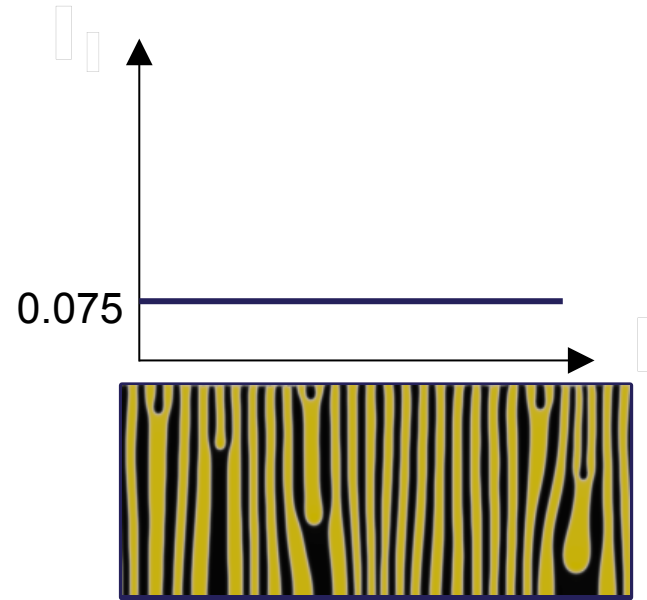


Source: [renata.com](http://renata.com)

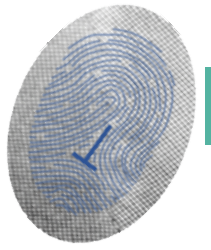
Thin film structure decides properties



# Microstructures from simple time-dependent protocols



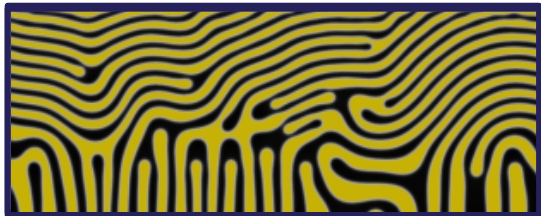
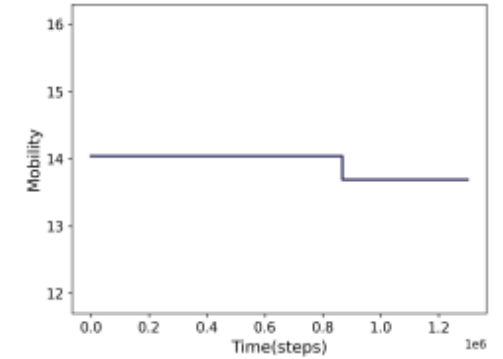
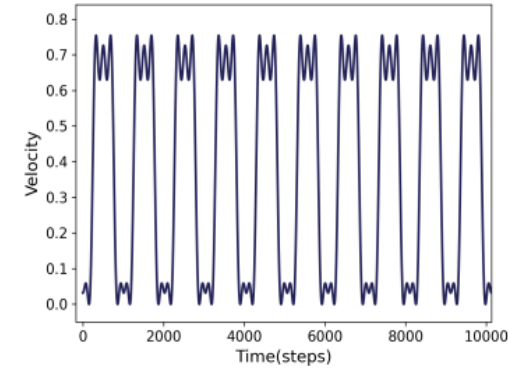
# Optimizing the deposition rate and the mobility



Target



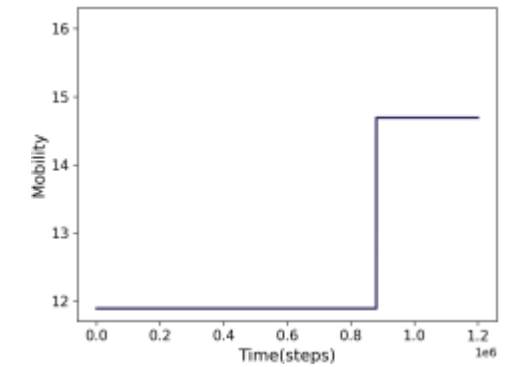
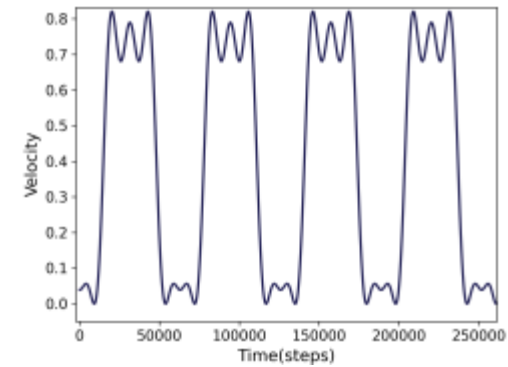
GA



Target



GA



# Phase field simulations of alloy deposition



$$F = \int \left\{ f_\phi + \frac{\kappa_\phi}{2} (\nabla \phi)^2 + s(\phi) \left( f_c + \frac{\kappa_c}{2} (\nabla c)^2 \right) \right\} d\Omega$$

Free energy of system

$$\frac{\partial c}{\partial t} = \nabla \cdot \left[ \mathbf{M}_c(\phi, c) \nabla \frac{\delta F}{\delta c} \right]$$

Evolution equation

$$\mathbf{M}_c(\phi, c) = \mathbf{M}^{bulk} + \mathbf{M}^{surf}$$

Surface mobility dominates microstructure

$$\mathbf{M}^{bulk} = \frac{1}{4} (2 - \phi) (1 + \phi)^2 \left[ h(c) \mathbf{M}_A^{bulk} + (1 - h(c)) \mathbf{M}_B^{bulk} \right]$$

$$\mathbf{M}^{surf} = e^{-\left(\frac{\phi}{\sigma^{surf}}\right)^2} \left[ h(c) \mathbf{M}_A^{surf} + (1 - h(c)) \mathbf{M}_B^{surf} - \mathbf{M}^{bulk} \right]$$

$$\frac{\partial \phi}{\partial t} = \nabla \cdot \left[ \mathbf{M}(\phi) \nabla \frac{\delta F}{\delta \phi} \right] + S(n(\phi))$$

Evolution equation with source term

$$\frac{\partial \rho}{\partial t} = \nabla \cdot [\mathbf{D}_\rho \nabla \rho] - \nabla \cdot [\rho \mathbf{v}] - S(n(\phi))$$

Vapor evolution equation

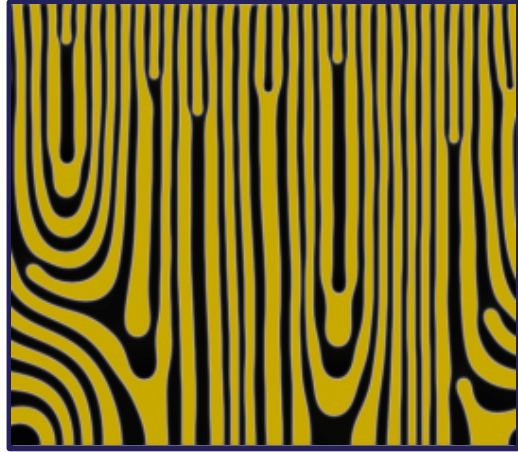
Phase field model simulates microstructure evolution for various deposition conditions



# Microstructures using constant deposition conditions



Deposition rate = 0.075



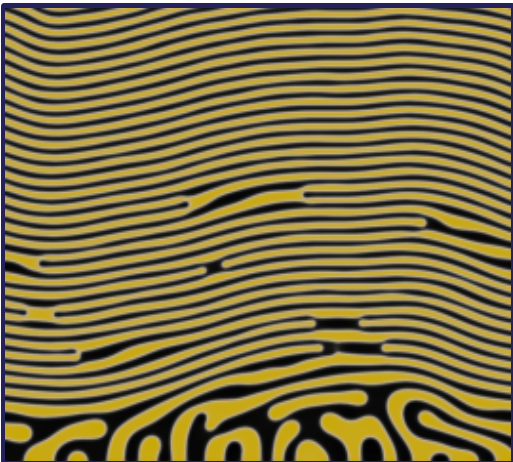
*Lateral concentration modulation (LCM)*

*Random concentration modulation (RCM)*

Deposition rate = 1.25



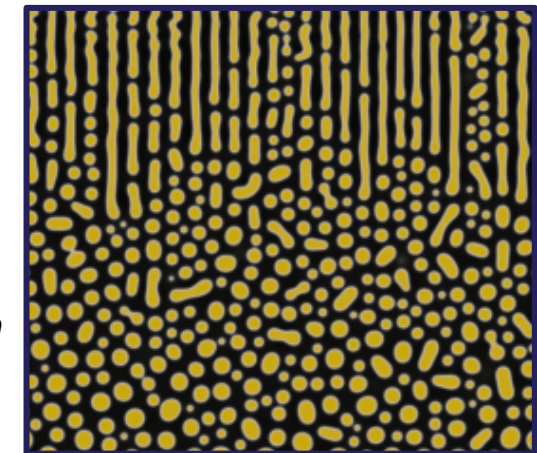
Deposition rate = 0.5



*Vertical concentration modulation (VCM)*

*Nanoprecipitate concentration modulation (NPCM)*

Deposition rate = 1.25



Competition between deposition and diffusion gives different microstructure regimes

$A_{65}B_{35}$