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Machine Learned Interatomic Potential Development for W-ZrC

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1444 Computational Multiscale

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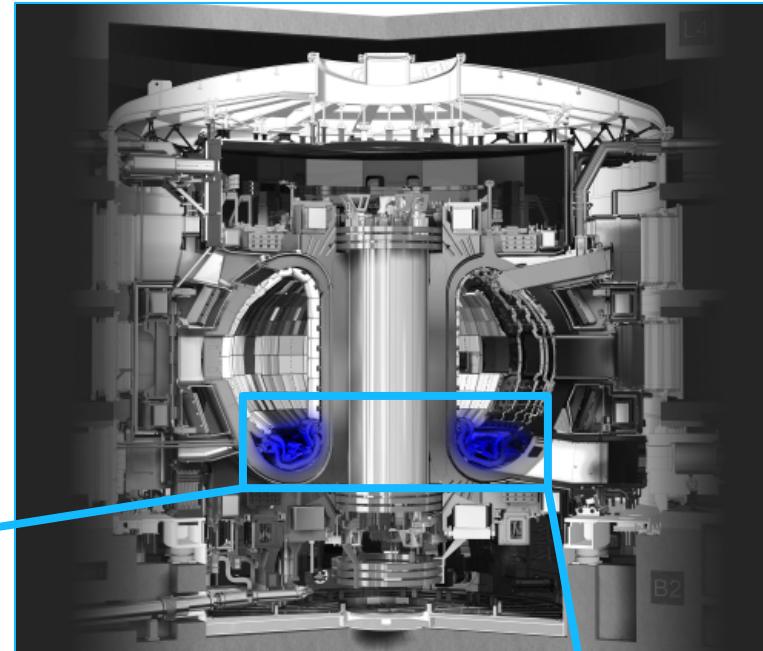


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Divertor Materials for ITER and Future Fusion Reactors



- The divertor controls the exhaust of waste gas and impurities
- Withstands the highest surface heat loads, up to 20 MW/m^2
- Tungsten (W) has optimal thermomechanical properties for a plasma-facing material, but has a high ductile-to-brittle transition temperature
- Strengthening W with **zirconium carbide (ZrC) dispersoids** can improve mechanical properties



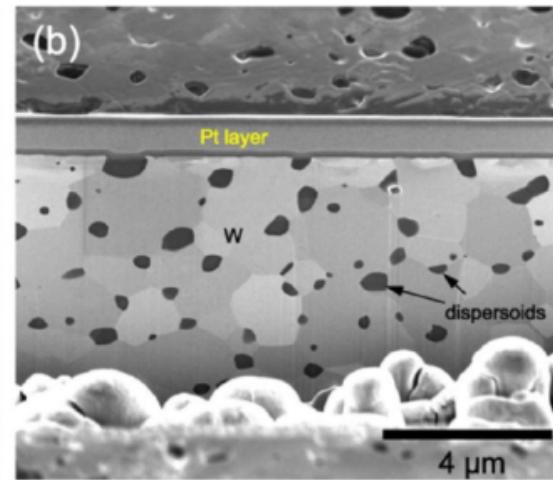
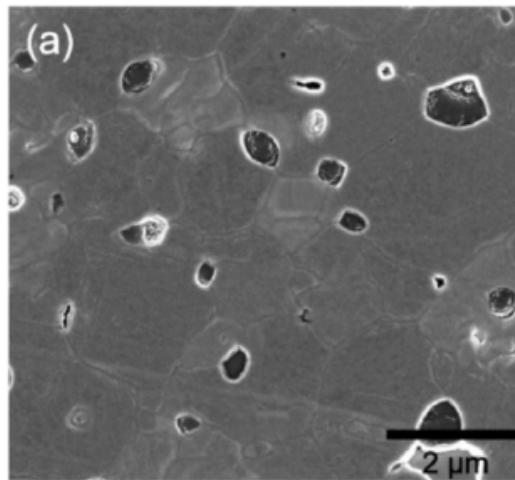
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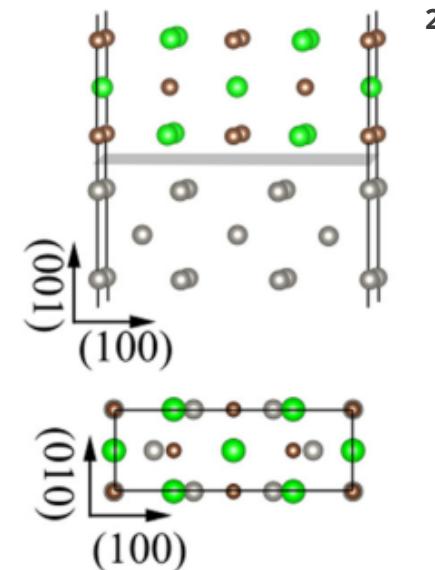
How can we investigate the mechanical properties of ZrC dispersoid-strengthened W?



- Plasma facing materials in ITER may experience temperatures > 2000 °C¹ which leads to W recrystallization and grain growth.
- W microstructural changes degrade strength, decrease fracture toughness, and enhance embrittlement.
- ZrC dispersoids have shown to improve ductility and limit grain growth.
- Mechanisms for these are still unknown and atomistic modeling can play a role in understanding these materials.



SEM of dispersoid strengthened W



DFT ~100 atoms

¹E. Lang et al. / JNM 545 (2021) 152613

²X. Zhang et al. / Appl. Surf. Science 499 (2020) 143995

The Spectral Neighbor Analysis Potential (SNAP) can map quantum data to a classical interatomic potential.



Model Form

- Each neighbor position, (r, θ, ϕ) , is mapped to a point, (θ_0, ϕ, θ) , on the unit 3-sphere.
- The basis can be described with bispectrum components, B_k^i .
- Fitting the linear coefficients, β_k , produces the SNAP potential:

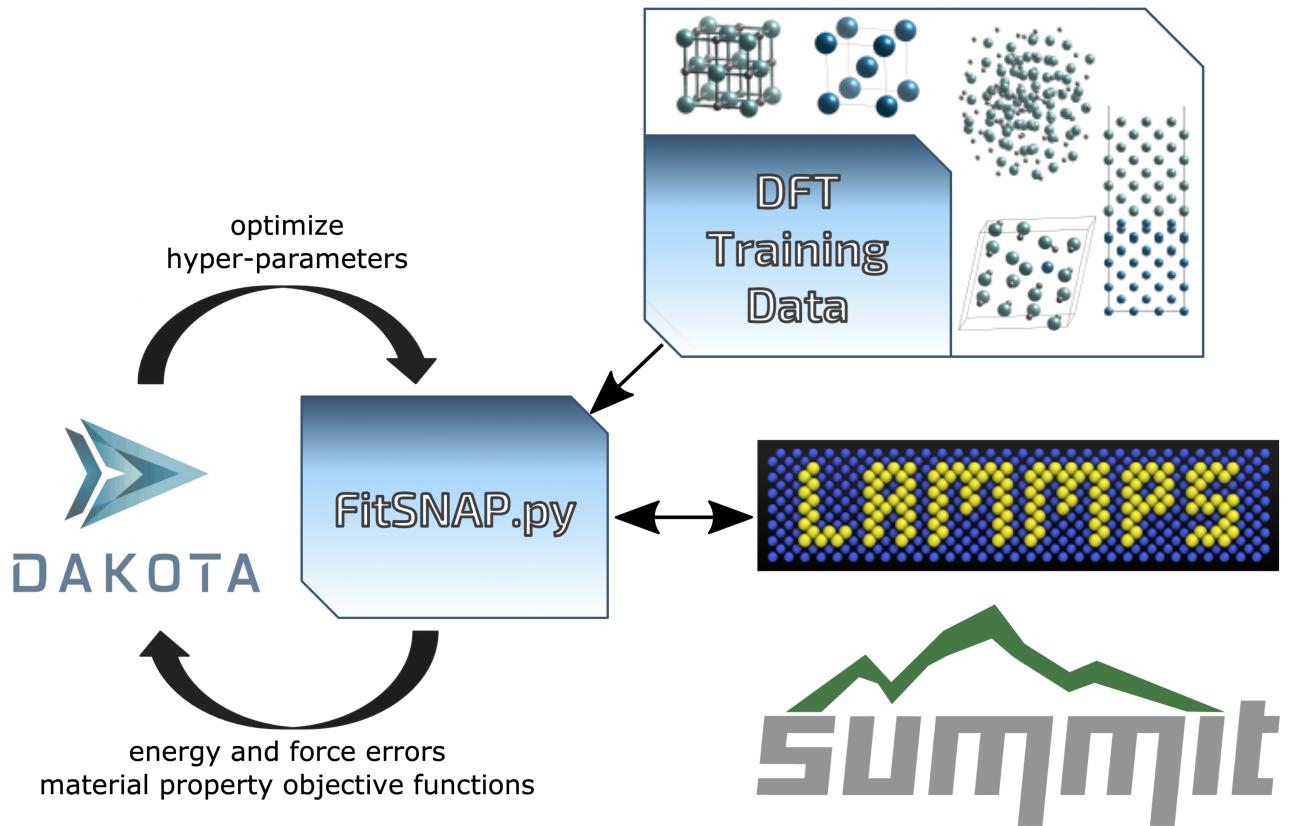
$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_k^i 0)$$

Linear Regression

$$\min(||\epsilon \cdot (D\beta - T)||^2)$$

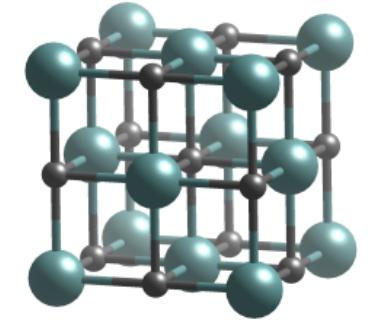
group weight descriptor prediction DFT training

Work flow

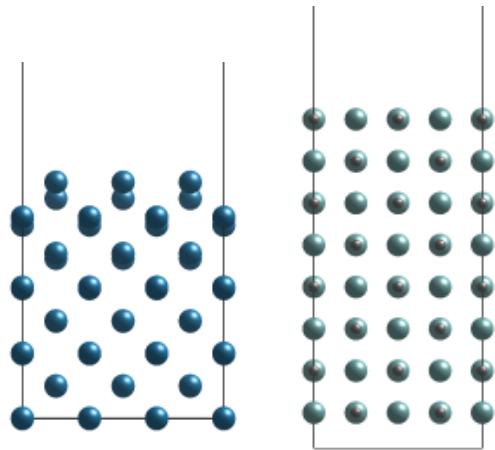
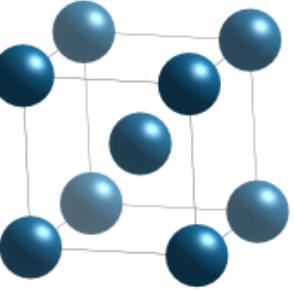


Code available: <https://github.com/FitSNAP/FitSNAP>

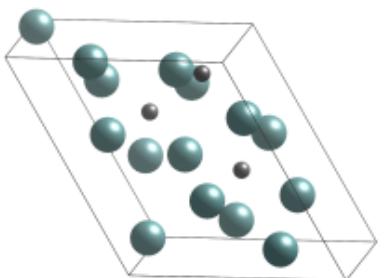
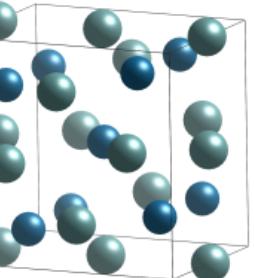
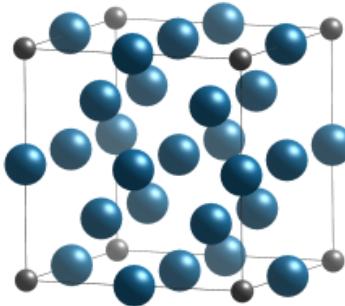
The SNAP W-ZrC potential is trained on a variety of first-principles data (~8,000 structures).



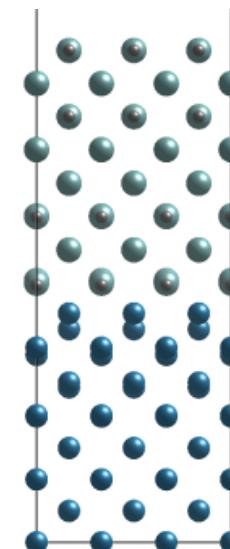
Bulk



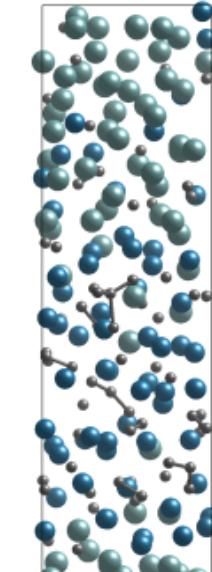
Surfaces



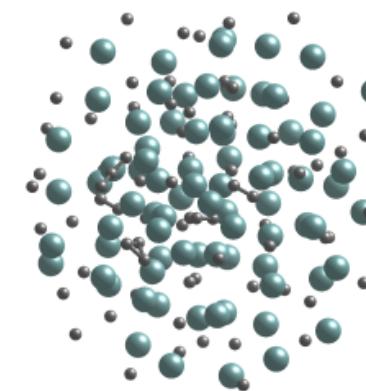
USPEX¹



Interfaces



Liquids



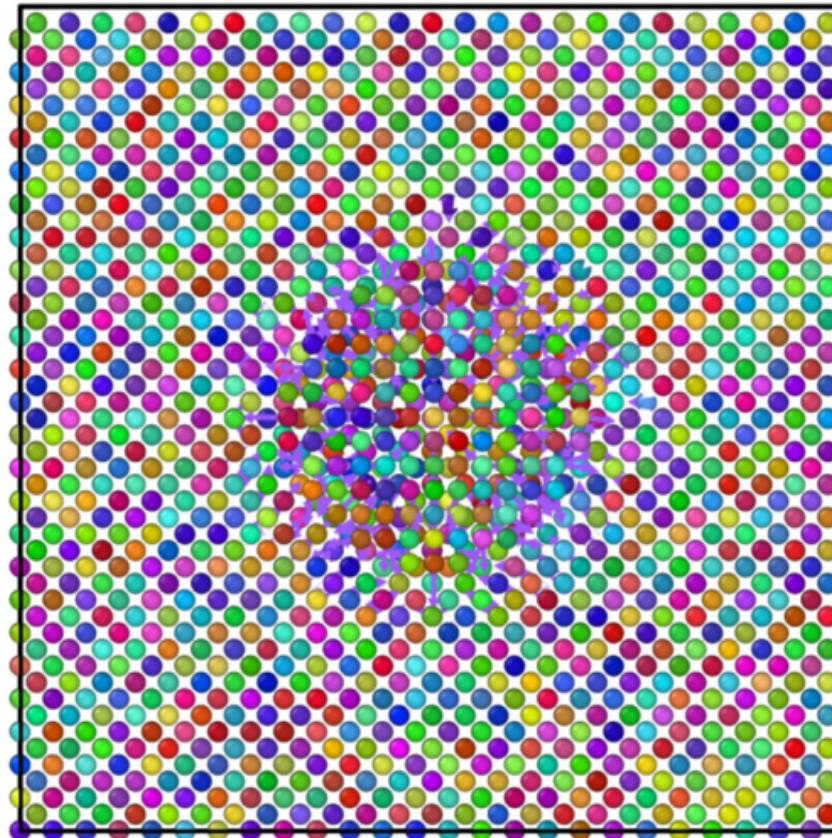
“active learning”



Candidate potentials need to be checked in NVT simulations for stability.

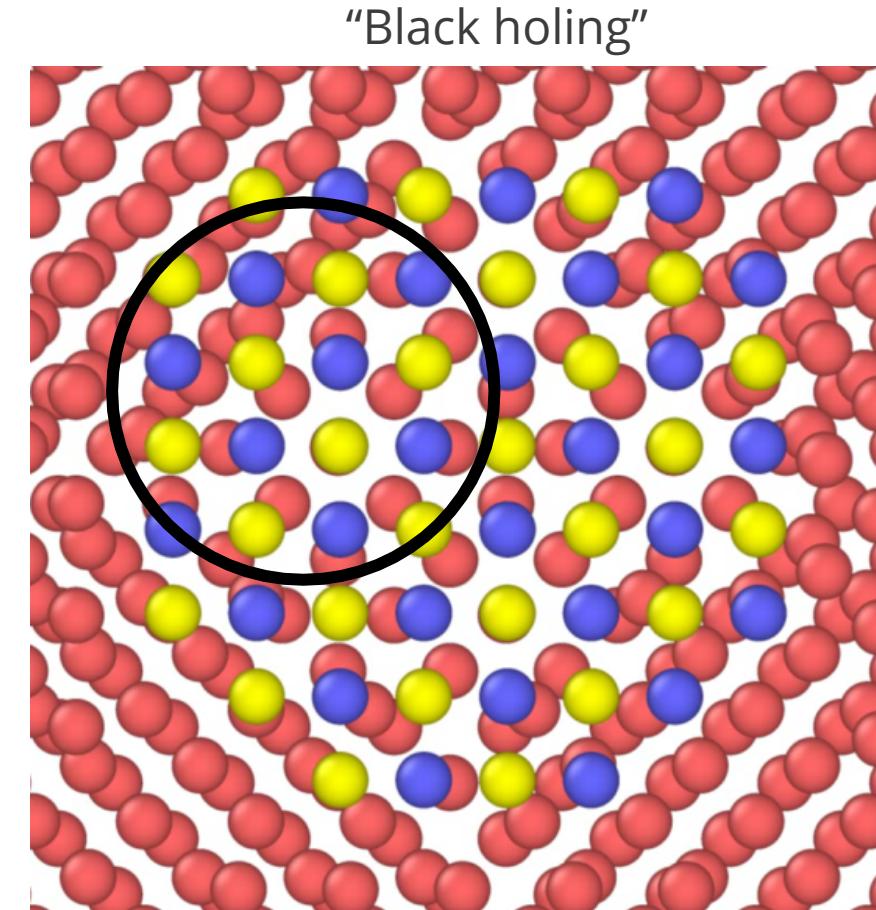


Cluster analysis
 $<2.2 \text{ \AA}$



$$\min(||\epsilon \cdot (D\beta - T)||^2)$$

- Increase group weight for forces on surface training data



- Add compression data
- Tune zbl overlay
 - Increase repulsive diameter
 - Decrease radial cutoff

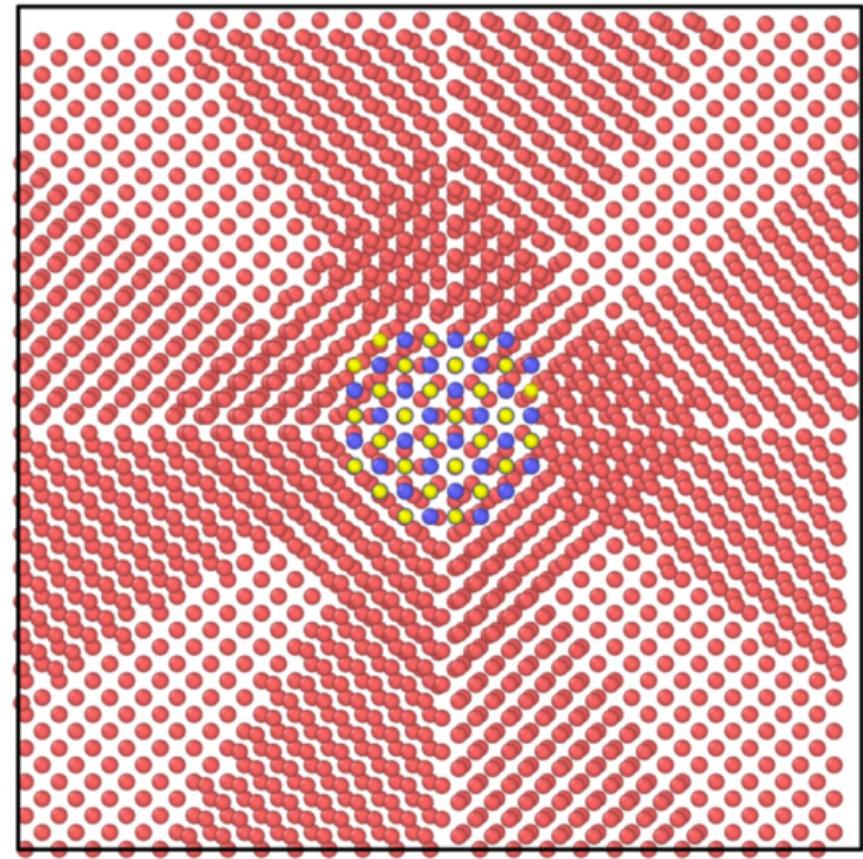
Genetic algorithm-generated SNAP potentials are narrowed down with MD tests.



- Potentials are tested on metrics including bulk modulus, B (GPa), and surface energies, E_{surf} (eV/Å²).

	B_W	B_{ZrC}	$E_{surf} W$ (100)	$E_{surf} W$ (110)	$E_{surf} ZrC$ (200)	$E_{surf} ZrC$ (110)
DFT	301.4	216.0	4.13	3.18	1.63	3.31
SNAP	299.8	210.9	3.57	2.67	1.23	2.44

- MD minimization and NVT test simulations are performed to check potential stability.
- Best-performing potential is selected for large-scale tests and experiments.

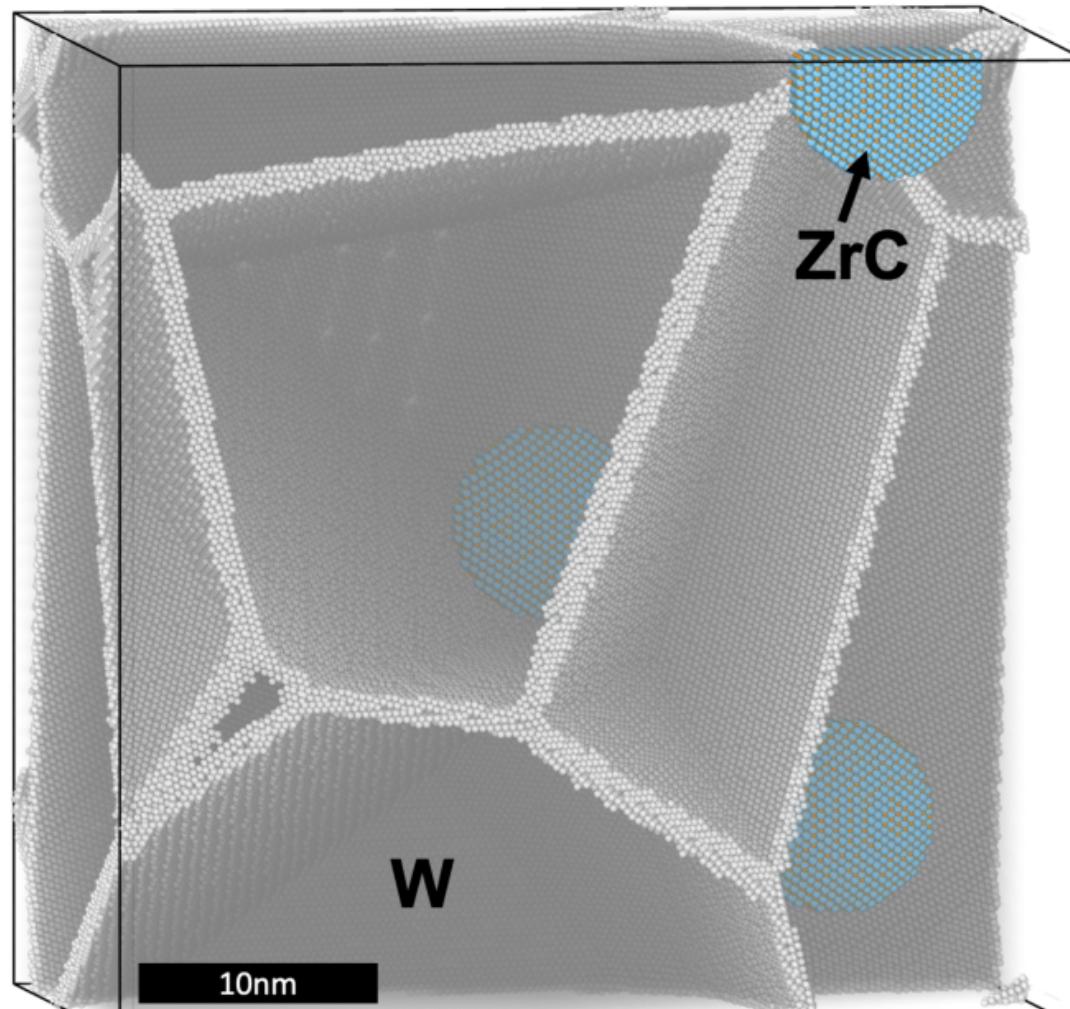


Example potential performance during minimize followed by NVT at 1000K.

Next steps: simulations on Summit (ORNL)



- Coherent interfaces
- Incoherent particles in W
- Material deformation at range of temperatures
 - 50 K (well below brittle--ductile transition)
 - 300 K (ambient conditions)
 - 1000 K (normal ITER operation)
 - 2500 K (approaching W melt)
- Other Summit simulations using SNAP:
 - Hydrogen (H) behavior in W in bulk and at interfaces
 - W-H SNAP: Mary Alice Cusentino
 - Radiation damage and recovery in refractory high-entropy alloys
 - Mo-Nb-Ta-Ti SNAP: Megan McCarthy

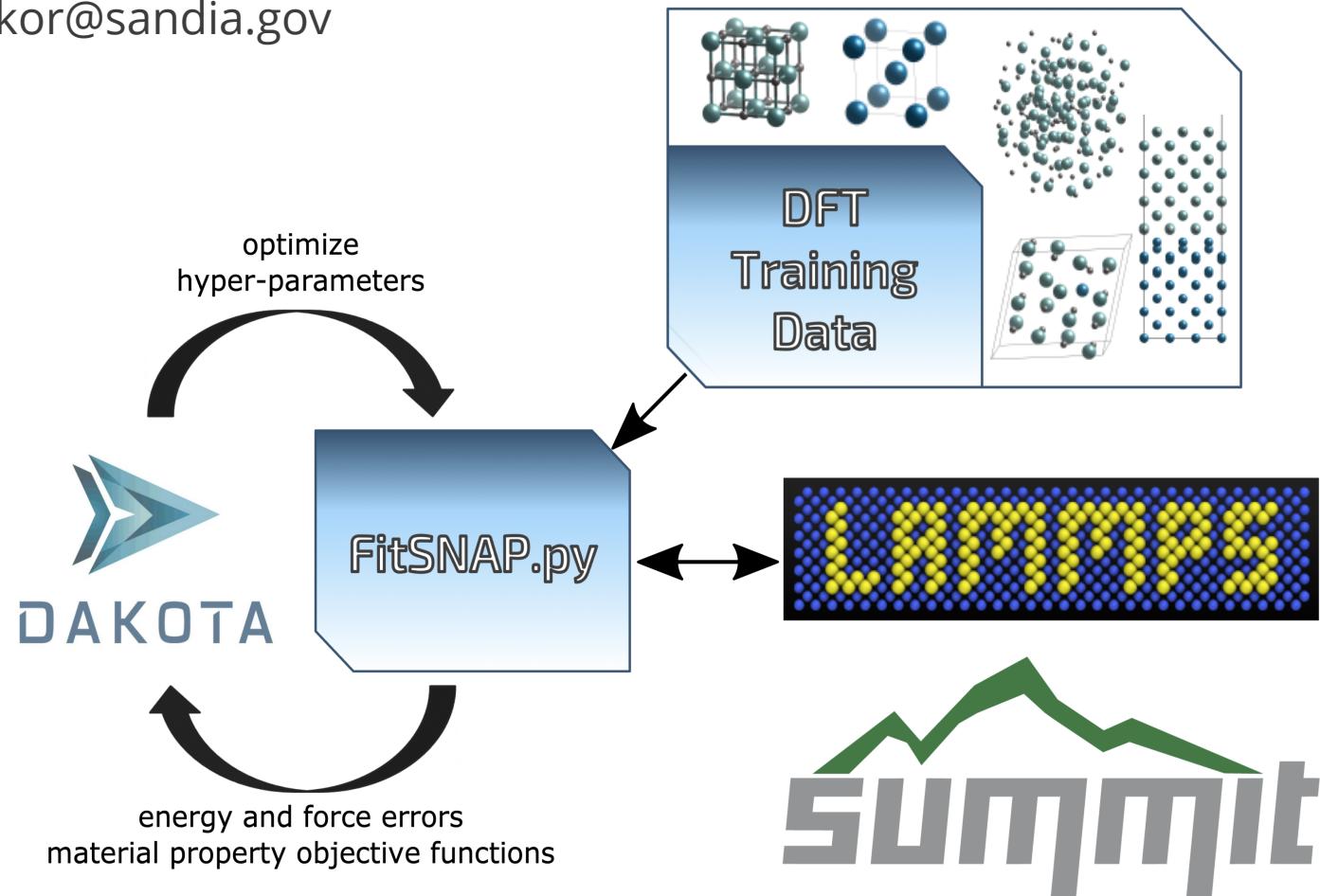


Example MD simulation geometry of ZrC dispersoid particles embedded in a polycrystalline W matrix

Thank you for your time.



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