

# Neutron Diffraction Probing Hydrogen in Monoclinic $\text{H}_2\text{VOPO}_4$

Yan Chen<sup>1,\*</sup>, Krystal Lee<sup>2</sup>, Ke An<sup>1</sup>, Dunji Yu<sup>1</sup>, Hui Zhou<sup>2</sup>, M. Stanley Whittingham<sup>2</sup>

<sup>1</sup> Neutron Scattering Division, Oak Ridge National Laboratory, Oak Ridge, TN 37831, USA

<sup>2</sup> NECCES, Binghamton University, Binghamton, NY 13902, USA

\* chenyl@ornl.gov

## Abstract

Light hydrogen atoms are resolvable with neutrons; however, the massive incoherent background inhibits the diffraction quality. To improve the signal, isotope treatment by deuteration becomes a prerequisite for a successful crystallographic understanding of hydrogen-containing materials by neutron diffraction. Thanks to the low-background and high-resolution time-of-flight neutron diffractometer, this work demonstrates a direct and successful measurement of high-quality neutron diffraction patterns of  $\text{H}_2\text{VOPO}_4$  powders, a precursor of a high-capacity alkali-ion battery cathode. The Rietveld refinement identifies the hydrogen coordinates, occupancy, and thermal parameters in the  $\text{H}_2\text{VOPO}_4$  lattice. The result highlights the unique capability of neutron diffraction for the structure characterization of hydrogen-containing materials, potentially without requiring costly and possibly artifact-inducing deuteration for neutron diffraction.

**Keywords:** Neutron diffraction; Hydrogen-containing material; Crystalline powders; Energy storage; Energy conversion; High resolution

## 1 Introduction

Probing hydrogen (H), the lightest element, in crystalline materials is rather challenging because of the lack of sensitivity in many structure characterizations such as x-rays, electrons etc. Non-destructive neutron diffraction, on the other hand, possesses a unique capability of detecting H. The distinguished negative coherent scattering length ( $-3.74$  fm) [1] not only contributes to the remarkable sensitivity of H, but also gives strong contrasts to differentiate from other elements that commonly have a positive value of scattering length. However, H simultaneously boosts the scattering background in neutron diffraction, due to the large values of both the incoherent scattering length ( $25.3$  fm) and the cross section ( $80.27$  barn) [1]. The high background negatively impacts the diffraction pattern quality and thus the crystallography analysis. The deuteration, substituting natural hydrogen with the costly isotope deuterium (D), is usually recommended in studying H-containing material via neutron diffraction. This isotope can help reduce the incoherent background (from  $25.3$  to  $4.04$  fm and from  $80.27$  to  $2.05$  barn) [1]. However, both the high cost and the limited availability raise considerations of this approach. The positive coherent scattering length of D also mitigates the contrast with many other elements

that also have close scattering length values. Last but not least, deuteration may change material structure in some instances [2].

This work demonstrates a direct measurement of a high-quality neutron diffraction pattern of a H-containing powder sample. The success is attributed to the high neutron flux, high d-space resolution, low instrument background, and high detector efficiency of the time-of-flight neutron diffractometer at the Spallation Neutron Source (SNS). Monoclinic  $\text{H}_2\text{VOPO}_4$ , which possesses about 22% hydrogen in atomic fraction, is selected as the model material in this study. It is an important precursor for the  $\epsilon$ - $\text{VOPO}_4$  synthesis, an attractive intercalation compound used as cathode material of alkali-ion batteries [3]. Its structure was reported to follow  $C2/c$  space group via X-ray diffraction (XRD), but the information of H in the lattice is missing [4,5] because of the little sensitivity of lightweight elements in XRD. The neutron powder diffraction in this work fills that gap by uncovering the H arrangement and vacancy in  $\text{H}_2\text{VOPO}_4$ .

## 2 Experiment

The monoclinic  $\text{H}_2\text{VOPO}_4$  powders were synthesized by combining  $\text{VCl}_3$  (Sigma-Aldrich, 97%) and  $\text{P}_2\text{O}_5$  (Sigma-Aldrich,  $\geq 98\%$ ) in 30 mL 190-proof ethanol. The solution was placed in a 4748 Type 125 mL PTFE-lined reactor (Parr Instrument Co.), and the reaction was set to 180 °C for 72 hours. The hydrothermal product was filtered and then dried at 65 °C overnight to yield the  $\text{H}_2\text{VOPO}_4$  powders. To form the  $\epsilon$ - $\text{VOPO}_4$  powders, the hydrogen of  $\text{H}_2\text{VOPO}_4$  was removed by annealing at 550 °C in flowing oxygen for 3 hours.

The synthesized powders were filled and sealed in vanadium cans with a 6 mm diameter for neutron powder diffraction. It was carried out at VULCAN diffractometer [6] at SNS at Oak Ridge National Laboratory. The 20Hz chopper and the high-resolution guide were configured. The incident neutron beam size was determined by the  $6 \times 12 \text{ mm}^2$  slits. Each powder sample was measured for one hour under the nominal 1.4MW power of SNS operation. The neutron events that were received from the  $^3\text{He}$  linear position sensitive detector bank at  $2\theta=90^\circ$  were reduced to a histogram. Under the same configuration, the standard Si powder sample and vanadium rod sample were measured for instrument calibration and incident spectrum normalization, respectively. Rietveld refinement including Fourier map calculation of the neutron diffraction pattern, was carried out using GSAS [7].

## 3 Results and discussion

A high-quality neutron diffraction pattern of  $\text{H}_2\text{VOPO}_4$  powders was successfully measured despite the challenge of the high H content. In Figure 1(a), the raw diffraction histograms are compared between  $\text{H}_2\text{VOPO}_4$  and  $\text{VOPO}_4$ . The strong incoherent scattering from the H atoms raises the background of  $\text{H}_2\text{VOPO}_4$  150% more than  $\text{VOPO}_4$ . Nevertheless, the coherent scattering that gives rise to the Bragg peaks of  $\text{H}_2\text{VOPO}_4$  is still pronounced, which does not seem to be weaker than those of  $\text{VOPO}_4$  in general. After the normalization to the incident beam spectrum, the high-resolution diffraction pattern, though with an elevated background, exhibits sharp and clear Bragg peaks to allow the crystallographic analysis (Figure 1(b)).

The Rietveld refinement over the neutron diffraction probes the lightweight H atoms at an 8f site of the initial structure model without H [4,5]. Figure 1(b) shows the full pattern fitting with least residuals. If the H was removed in the fitting, the significant mismatch would be observed (Figure 1(b) inset), which indicates the pronounced contribution of H to the Bragg peaks. It thus illustrates the high fidelity of probing this lightweight element via neutron diffraction. This refinement determined the lattice parameters, atoms' positions in the lattice, site fractional occupancies, and anisotropic displacement parameters in  $\text{H}_2\text{VOPO}_4$  (Table 1). It is worth noting that no fitting constraints were applied to those variables, except for the V atom because its coherent scattering length is so weak [1] to make it not sensitive to neutron diffraction.

The crystal structure resulting from the refinement is illustrated in Figure 2. The framework of  $\text{H}_2\text{VOPO}_4$  composes of connecting  $\text{PO}_4$  tetrahedra and  $\text{VO}_6$  octahedra. A  $\text{PO}_4$  shares the four corners (two O(1) and two O(2)) with the four neighboring  $\text{VO}_6$ , respectively. A  $\text{VO}_6$  also shares the two O(3) corners with another two  $\text{VO}_6$  octahedra, respectively. The two hydrogen atoms bond with the O(3). In addition, each H atom could have a weaker hydrogen bond with the O(1) of a neighboring  $\text{VO}_6$ . This structure of  $\text{H}_2\text{VOPO}_4$  is found to be consistent with  $\text{MnPO}_4 \cdot \text{D}_2\text{O}$  [8], which was determined using deuteration.

The properties of the bonds are revealed from the structure refinement. In the  $\text{PO}_4$  tetrahedron, the bond lengths of P–O(1) and P–O(2) are consistent. It indicates the strong covalent bonds with  $\text{sp}^3$  hybridization at the P(V) ion. Differently, the  $\text{VO}_6$  octahedron is stretched along V–O(3) in comparison to V–O(1) and V–O(2). It results from the Jahn-Teller effects of the V(III) ion that has 2 electrons at the 3d orbitals. This distortion of the octahedron is also correlated to the bond to H. Due to the existence of the O(3)–H bond, the V–O(3) bond is weakened and thus elongated. Similarly, the  $\text{H} \cdots \text{O}(1)$  hydrogen bond also influences the bonds of O(1) while the O(2) is nearly screened from H atoms. Therefore, it is observed that both the V–O(1) and P–O(1) bonds have slightly larger lengths than V–O(2) and P–O(2), respectively.

In the  $\text{H}_2\text{VOPO}_4$ , the O(3)–H bond length is 1.013 Å, and the H–O(3)–H angle is 108.7°. The geometry of H–O–H group is slightly changed from that in the  $\text{H}_2\text{O}$  molecule, compared to the 0.9572 Å bond length and the 104.52° bond angle in  $\text{H}_2\text{O}$  [9]. Both values are larger in the  $\text{H}_2\text{VOPO}_4$ . This may be attributed to the V–O(3) bond, which attracts the electron cloud around O(3) to weaken the O–H interaction and enhance the repulsion between the two H atoms.

The vacancies of the lightweight element H and O are probed via the refinement. Without constraints, it is calculated to have a small amount of vacancies at the H and the O(3) sites while the O(1) and O(2) sites are fully occupied within the fitting errors. The loss of H and O follows the atom ratio of a water molecule, showing an equivalued formula  $\text{VPO}_4 \cdot 0.945\text{H}_2\text{O}$ . The variation of  $\text{H}_2\text{O}$  content is commonly observed via thermogravimetry, depending on the synthesis process [4]. Therefore, the stoichiometry changes do not lead to an oxidation of V, and the O(1) and O(2) bonding to V and P are well maintained.

The anisotropic displacement parameters ( $U_{ij}$ ) are calculated for each atom except V. The light H exhibit relatively large  $U_{ij}$  values with strong anisotropy. It may be contributed by the dynamic vibration of H that has a single bond to O(3) and also be attributed to some possible static local displacement in disorder [10]. The refined coordinates and the  $U_{ij}$  give the average state of the H atom.

## 4 Conclusion

The high-quality time-of-flight neutron diffraction pattern was directly measured for the  $\text{H}_2\text{VOPO}_4$  powder sample. The site position, occupancy, and the anisotropic thermal vibration of the hydrogen atoms were probed. The presence of H was correlated to the distortions of  $\text{VO}_6$  octahedron and the  $\text{PO}_4$  tetrahedron. This work successfully demonstrated the practice of probing H in the lattice at the advanced time-of-flight neutron diffractometer, without the need to impose additional material treatments like deuteration. The characterization exhibits potential extensions on studying a general catalogue of H-containing (small to medium amount) crystalline compounds in the form of powders or polycrystals, such as hydrides, hydroxides, hydrate, and so on, for engineering and energy applications.

## 5 Acknowledgements

This research used resources at the Spallation Neutron Source (SNS), a DOE Office of Science User Facility operated by the Oak Ridge National Laboratory. This work was also supported by LG Energy Solutions under award number 92617.

## 6 References

- [1] V.F. Sears, Neutron scattering lengths and cross sections. *Neutron News*, 3 (1992) 26-37.
- [2] Z. Tun, R.J. Nelme, W.F. Kuhs, R.F.D. Stansfield, A high-resolution neutron-diffraction study of the effects of deuteration on the crystal structure of  $\text{KH}_2\text{PO}_4$ , *J. Phys. C: Solid State Phys.* 21 (1988) 245-258.
- [3] M.S. Whittingham, Y. Song, S. Lutta, P.Y. Zavalij, N.A. Chernova, Some transition metal (oxy)phosphates and vanadium oxides for lithium batteries, *J. Mater. Chem.* 15 (2005) 3362–3379.
- [4] J.T. Vaughey, W.T.A. Harrison, A.J. Jacobson, D.P. Goshorn, J.W. Johnson, Synthesis, Structure, and Properties of Two New Vanadium (III) Phosphates:  $\text{VPO}_4 \cdot \text{H}_2\text{O}$  and  $\text{V}_{1.23}(\text{PO}_4)(\text{OH})_{0.69}(\text{H}_2\text{O})_{0.31} \cdot 0.33\text{H}_2\text{O}$ , *Inorg. Chem.* 33 (1994) 2481-2487.
- [5] A. El Badraoui, J.Y. Pivan, M. Maunaye, O. Pena, M. Louer, D. Louer, Order-disorder phenomena in vanadium phosphates. Structures and properties of tetragonal and monoclinic  $\text{VPO}_4(\text{H}_2\text{O})$ , *Ann. Chim. Sci. Mat.* 23 (1998) 97-101.
- [6] K. An, Y. Chen, A.D. Stoica, VULCAN: A “hammer” for high-temperature materials research, *MRS Bull.* 44 (2019) 878-885.
- [7] A.C. Larson, R. B. von Dreele, General structure analysis system (GSAS), Los Alamos National Laboratory Report LAUR 86–748 (2004).
- [8] A.G. Miguel, J.P. Attfield, S. Bruque, F. Palacio, Magnetic structures of  $\text{MnPO}_4 \cdot \text{D}_2\text{O}$  and  $\text{MnAsO}_4 \cdot \text{D}_2\text{O}$  from time-of-flight neutron powder diffraction data, *J. Mater. Chem.* 2 (1992) 501-505.
- [9] M. Huš, T. Urbic, Strength of hydrogen bonds of water depends on local environment, *J. Chem. Phys.* 136 (2012) 144305.
- [10] Y. Chen, Y. Cheng, J. Li, M. Feygenson, W.T. Heller, C. Liang, K. An, Lattice-Cell Orientation Disorder in Complex Spinel Oxides, *Adv. Energy Mater.* 7 (2017) 1601950.

## 7 Figure Captions

Figure 1 (a) As-collected neutron diffraction histograms for  $\text{H}_2\text{VOPO}_4$  and  $\text{VOPO}_4$ . (b) Normalized neutron diffraction pattern and Rietveld refinement of  $\text{H}_2\text{VOPO}_4$ . (The inset shows a refinement without hydrogen contribution while the parameters of the other atoms are unchanged.)

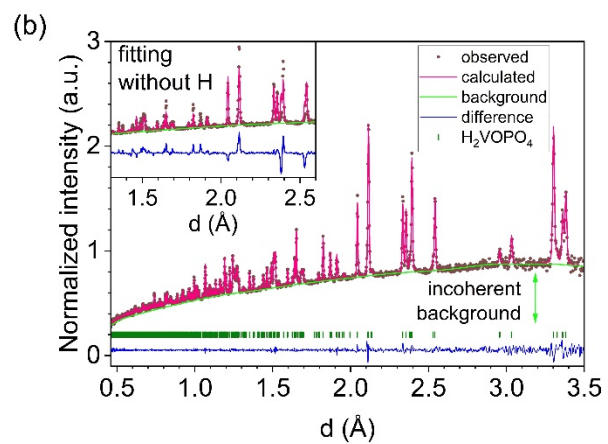
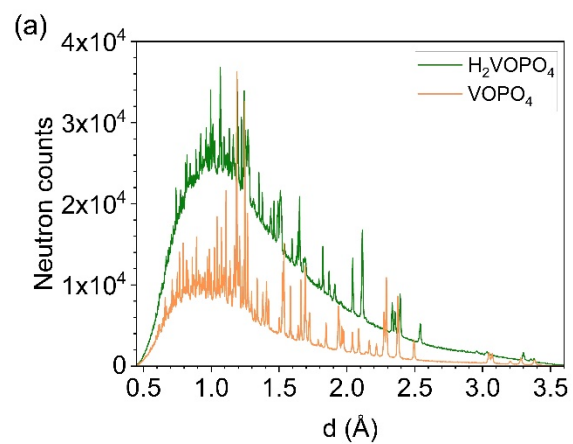
Figure 2 The crystal structure and bond lengths of  $\text{H}_2\text{VOPO}_4$  based on neutron diffraction.

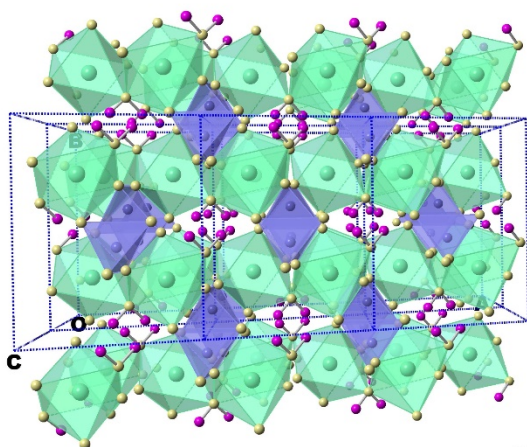
## 8 Tables

Table 1 Crystal structure of  $\text{H}_2\text{VOPO}_4$  (Space group:  $C2/c$ .  $a = 6.7093(2)\text{\AA}$ ;  $b = 7.7895(2)\text{\AA}$ ;  $c = 7.3565(1)\text{\AA}$ ;  $\beta = 115.308(2)^\circ$ ).

## 9 Supplementary data

Crystallographic Information File (CIF) of  $\text{H}_2\text{VOPO}_4$ .





V–O(1): 1.983(2)Å  
 V–O(2): 1.929(2)Å  
 V–O(3): 2.148(1)Å  
 P–O(1): 1.535(3)Å  
 P–O(2): 1.528(3)Å  
 H–O(3): 1.013(5)Å  
 H...O(1): 1.679(5)Å

