

Effects of spin-orbit coupling and very large supercells on the description of acceptor states in CdTe

Intuon Chatratin, Igor Evangelista, and Anderson Janotti

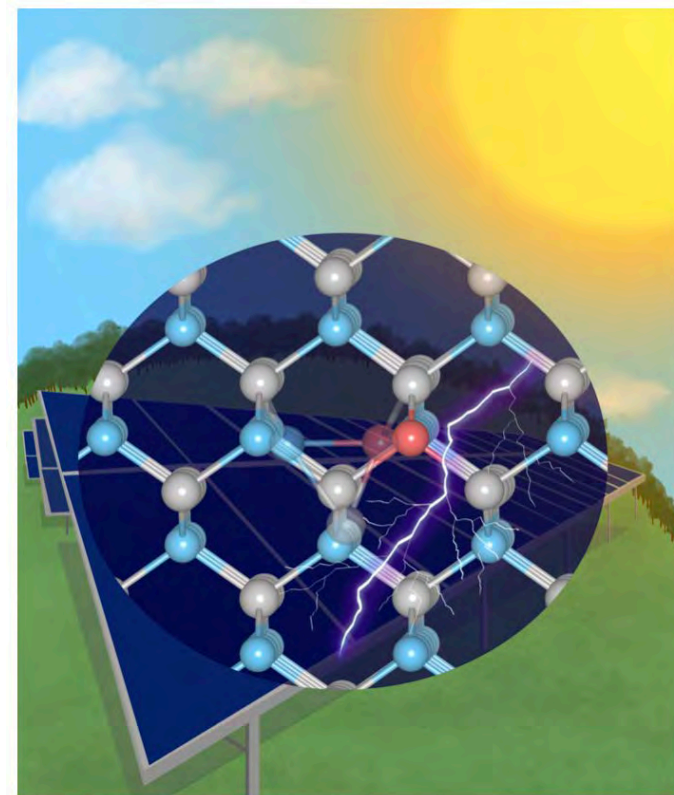
Department of Materials Science and Engineering
University of Delaware

Email: janotti@udel.edu

<https://sites.udel.edu/mseg-janotti-group>



SOLAR ENERGY
TECHNOLOGIES OFFICE
U.S. Department Of Energy



Award # DE-EE0009344

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International Conference on Defects
in Semiconductors

**September
10-15, 2023**

Rehoboth Beach, Delaware

Topics: Theory and experiments on
Wide-band-gap oxides and nitrides
Quantum defects, qubits, single-photon emitters
Conventional IV-IV, III-V, and II-VI semiconductors
Complex oxide and halide perovskites
Electronics, optoelectronics, photovoltaic, thermoelectric, magnetic,
ferroelectric, and spintronic materials, radiation detectors
2D materials

...

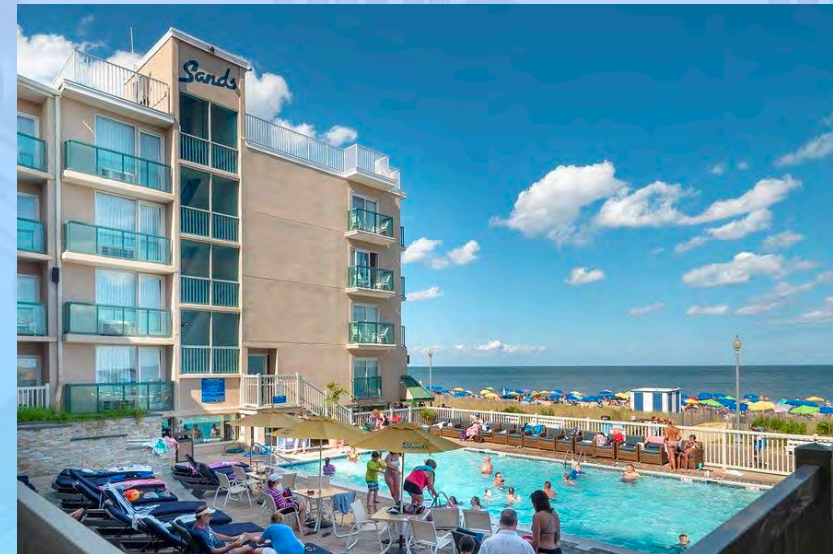
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**Abstract submission
will be open soon**

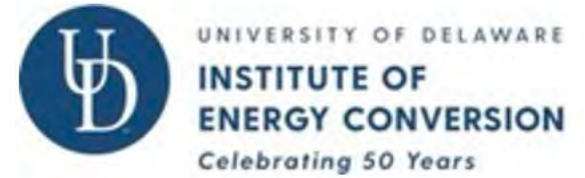


For more information:



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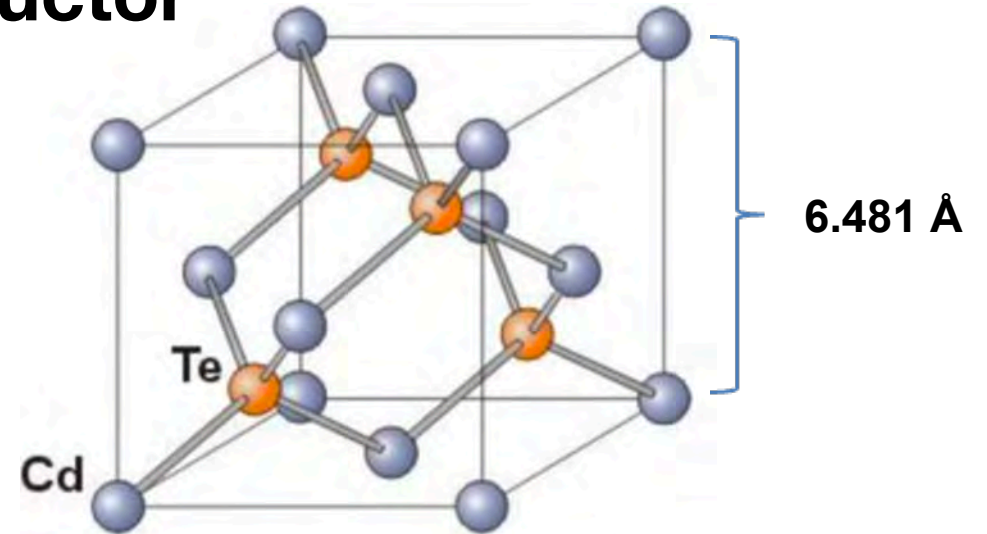


CdTe semiconductor

member of the II-VI family

zinc blende crystal structure

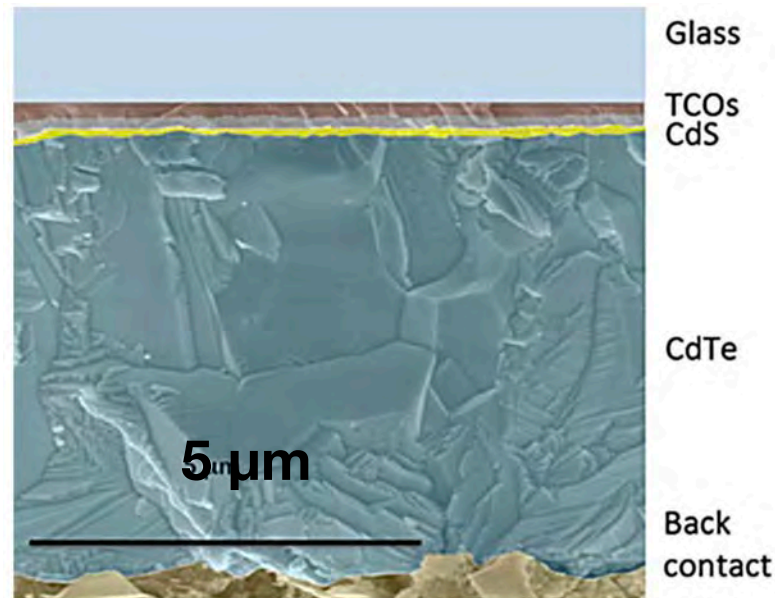
direct band gap = 1.5 eV



single crystals



thin films vapor transport deposition



Applications:

thin film solar cells

infrared optical windows/lenses

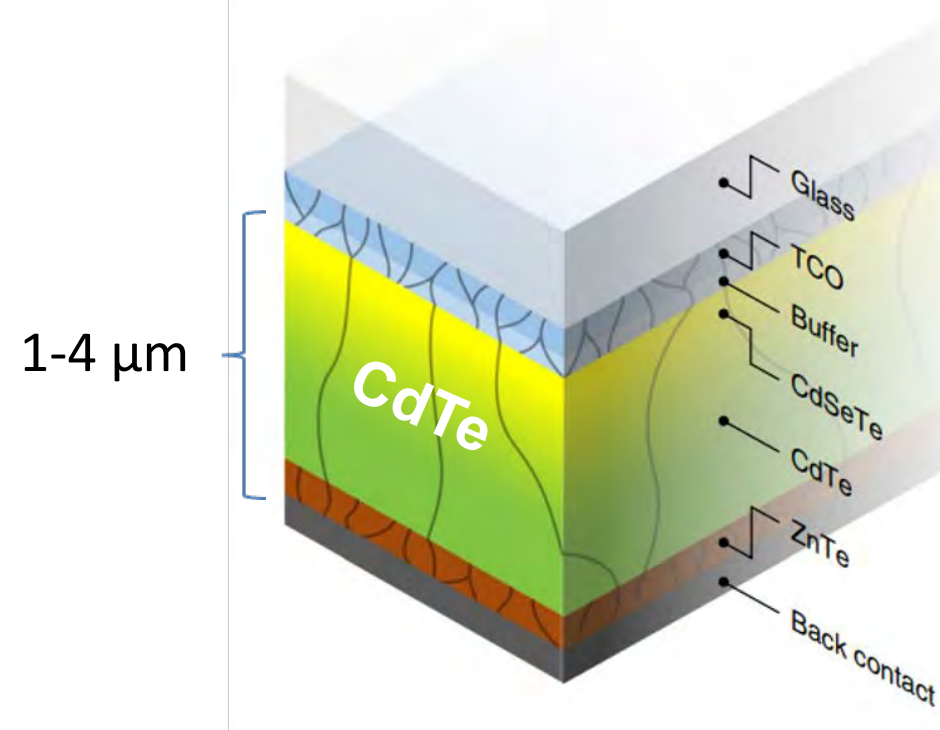
electro-optic modulators

scintillators

Matt Reese (NREL)

CdTe in solar cells

thin-film photovoltaic technology



Metzger *et al.* Nat. Energy **4**, 837 (2019)

record efficiency in the lab cells: 22.3%

Polycrystalline

→ grain boundaries

vapor transport deposition
→ low production cost

**Commercially
available**

19.3%
HIGH BIN EFFICIENCY

30YR
LINEAR PERFORMANCE
WARRANTY

Series 7

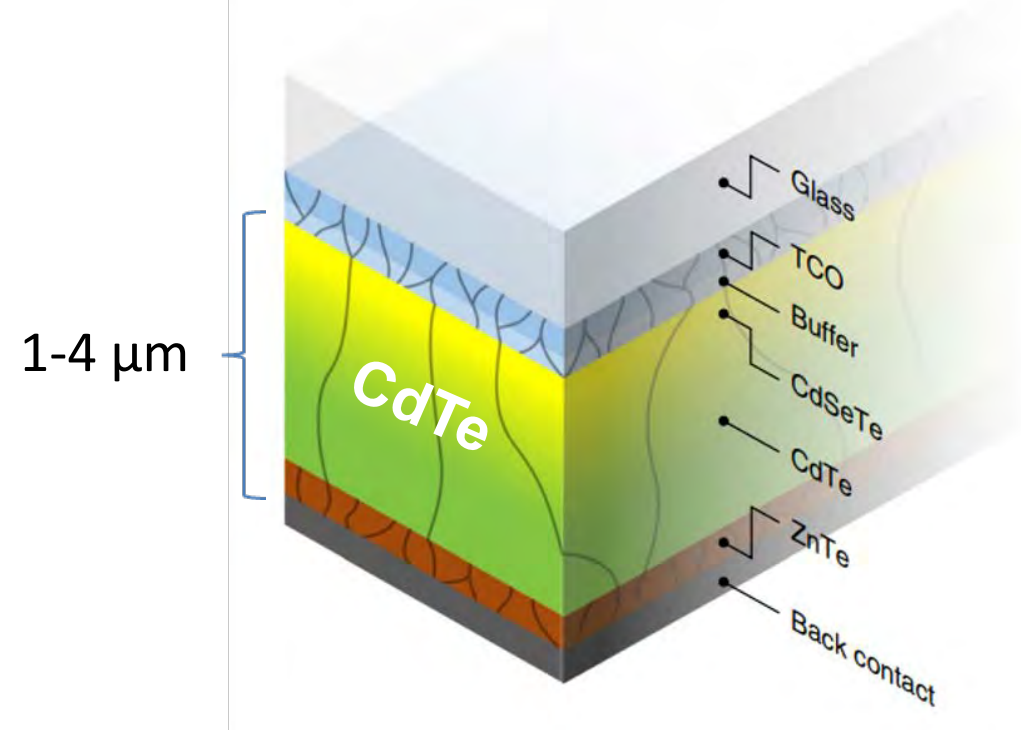
Made using same process to fabricate modules



CdTe in solar cells

thin-film photovoltaic technology

Typical minority-carrier device



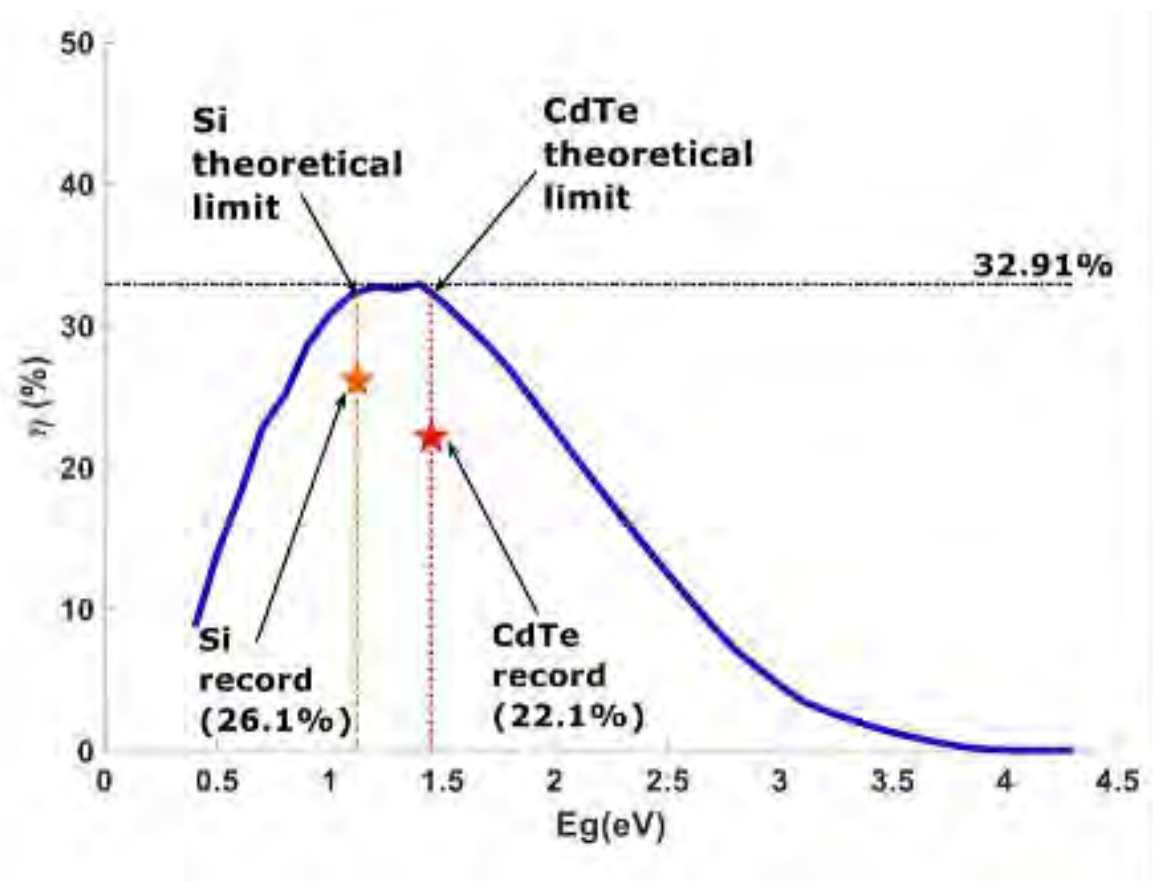
Metzger *et al.* Nat. Energy **4**, 837 (2019)

Current limitations:

- low open-circuit voltage V_{oc} ($\sim 0.8\text{-}0.9$ eV) $\ll E_g$ (1.5 eV)
- undoped, low hole concentrations $\sim 10^{14}$ cm $^{-3}$
- layers doped with As show very low doping efficiency
 - [As] $\sim 10^{18}$ cm $^{-3}$, [holes] $\sim 10^{16}$ cm $^{-3}$
where do the dopants go?
- short minority carrier lifetimes
- polycrystalline films
 - carrier recombination in the bulk
and at grain boundaries

CdTe solar cells - room for improvement

Current efficiencies are well below the theoretical limit of 33%



Device modeling indicate that efficiency of 25% can be achieved if:

hole concentration $> 10^{16} \text{ cm}^{-3}$
while keeping everything else the same

carrier lifetime $\geq 100 \text{ ns}$,
interface recombination velocity $\leq 1000 \text{ cm/s}$

Burst *et al.*, Nature Energy **1**, 16015 (2016).

Kanevce, *et al.*, J. Appl. Phys. **121**, 214506 (2017).

Barbato *et. al.* J. Phys. D. Appl. Phys. **54**, 333002 (2021)

Back to the basics of doping CdTe

Periodic table of the elements

group	1*											13	14	15	16	17	18
period	1											5	6	7	8	9	10
1	1											B	C	N	O	F	He
2	3	4											6	7	8	9	10
3	11	12											13	14	15	16	17
4	19	20	21	22	23	24	25	26	27	28	29	30	31	32	33	34	35
5	37	38	39	40	41	42	43	44	45	46	47	48	49	50	51	52	53
6	55	56	57	72	73	74	75	76	77	78	79	80	81	82	83	84	85
7	87	88	89	104	105	106	107	108	109	110	111	112	113	114	115	116	117
	Fr	Ra	Ac	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Nh	Fl	Mc	Lv	Ts

Back to the basic aspects of doping CdTe

Periodic table of the elements

period	group 1*	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18
1	1 H	2																2 He
2	3 Li	4 Be											5 B	6 C	7 N	8 O	9 F	10 Ne
3	11 Na	12 Mg											13 Al	14 Si	15 P	16 S	17 Cl	18 Ar
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe
6	55 Cs	56 Ba	57 La	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn
7	87 Fr	88 Ra	89 Ac	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Cn	113 Nh	114 Fl	115 Mc	116 Lv	117 Ts	118 Og
lanthanoid series 6	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb	71 Lu				
actinoid series 7	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No	103 Lr				

For p-type doping:

Look to the left of Cd or Te

Try and minimize chemical and size mismatches

Cu, Ag on the Cd site

P, As, or Sb on the Te site

*Numbering system adopted by the International Union of Pure and Applied Chemistry (IUPAC). © Encyclopædia Britannica, Inc.

Typical experimental data on doping p-type of CdTe

Cu-doped CdTe absorber layers $< 10^{15} \text{ cm}^{-3}$

+ stability issues \rightarrow Cu interstitials are highly mobile

Corwine *et al.*, *Sol. Energy Mater. Sol. Cells* 82, 481 (2004)

Grecu, *et al.*, *J. Appl. Phys.* 88, 2490 (2000)

Burst *et al.*, *APL Mater.* 4, 116102 (2016)

As, P, Sb doping: $10^{15} - 10^{16} \text{ holes/cm}^{-3}$

very low doping efficiency, $[\text{free carrier}] \ll [\text{dopant}]$

highly compensated

+ short carrier lifetimes

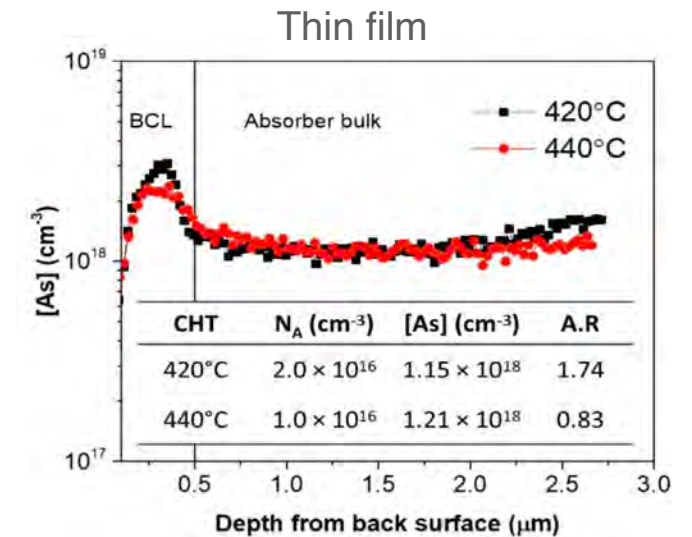
McCandless *et al.*, *IEEE J. Photovolt.* 9, 912 (2019)

Metzger *et al.*, *Nature Energy*, 4 837 (2019)

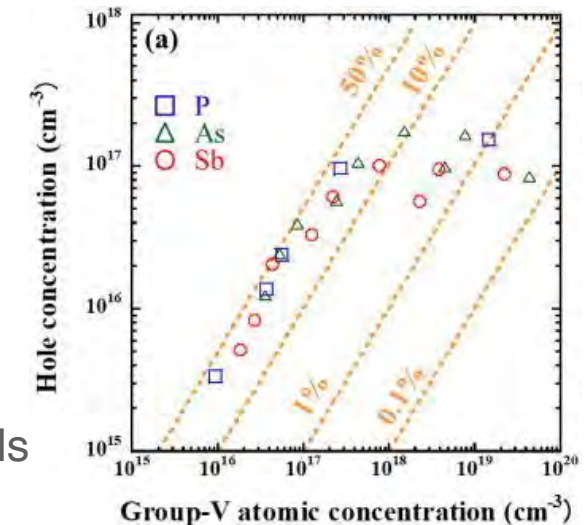
Kartopu *et al.*, *Sol. Energy Mater. Sol. Cells* 194, 259 (2019)

Source of compensation unknown!

Dopants in the grain boundaries?



Oklobia *et al.* *IEEE J. Photovolt.* **12**, 1296 (2022)



Single crystals

Nagaoka *et al.*, *Appl. Phys. Lett.* **116**, 132102 (2020)

Sb, As and P doping in CdTe single crystals

Temperature dependent Hall data

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Partially compensated acceptors

$$p = -A + \sqrt{A^2 + \frac{N_V}{2} (N_A - N_D) \exp\left(-\frac{E_a}{k_B T}\right)}$$

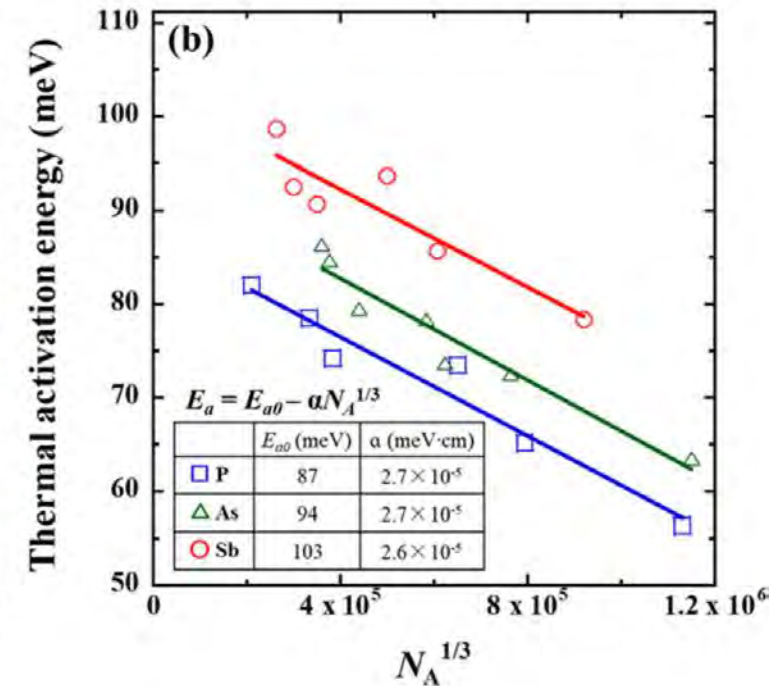
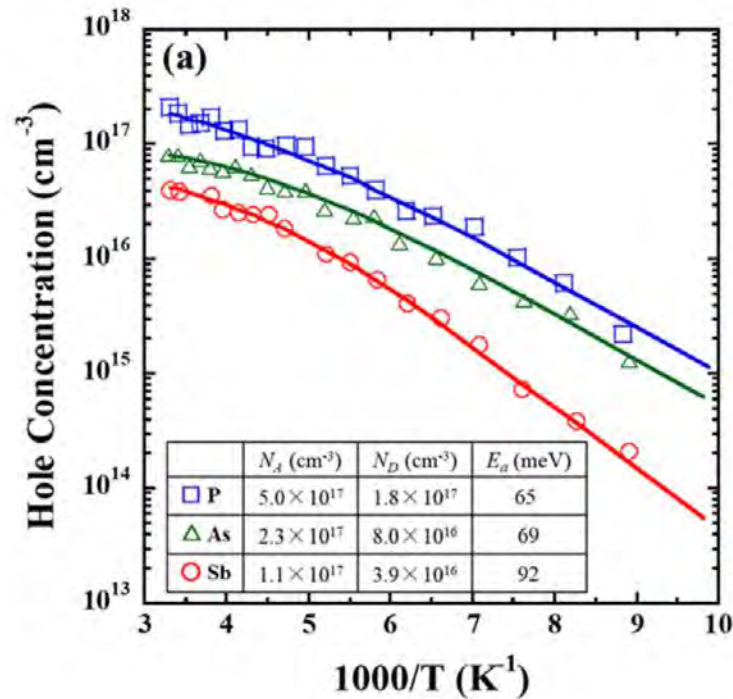
$$A = \frac{1}{2} \left[N_D + \frac{N_V}{2} \exp\left(-\frac{E_a}{k_B T}\right) \right]$$

Blakemore, Semiconductor statistics
(Courier Corp.,2002)

Data fit to obtain E_a , N_A ,...

	N_A (cm ⁻³)	N_D (cm ⁻³)	E_a (meV)
□ P	5.0×10^{17}	1.8×10^{17}	65
△ As	2.3×10^{17}	8.0×10^{16}	69
○ Sb	1.1×10^{17}	3.9×10^{16}	92

$$E_a = E_{a0} - \alpha N_A^{1/3}$$



	E_{a0} (meV)	α (meV·cm)
□ P	87	2.7×10^{-5}
△ As	94	2.7×10^{-5}
○ Sb	103	2.6×10^{-5}

dilute limit

What do we know about acceptor impurities in CdTe from theory

$$\text{Te}_i \xrightarrow{0.57} (0/-2)$$

DFT-LDA
LAPW, non-relativistic
Supercells with 32 atoms

				Bi _{Te}	$\frac{0.30}{0.23}$	
				Sb _{Te}		
V _{Cd}	$\frac{0.21}{0.13}$	(-2/-)	Cu _{Cd}	$\frac{0.22}{0.20}$	(0/-)	
			Au _{Cd}			
			Ag _{Cd}	$\frac{0.15}{0.02}$		
			Na _{Cd}			
				As _{Te}	$\frac{0.10}{0.05}$	(0/-)
				P _{Te}		
				N _{Te}	$\frac{0.01}{0.01}$	

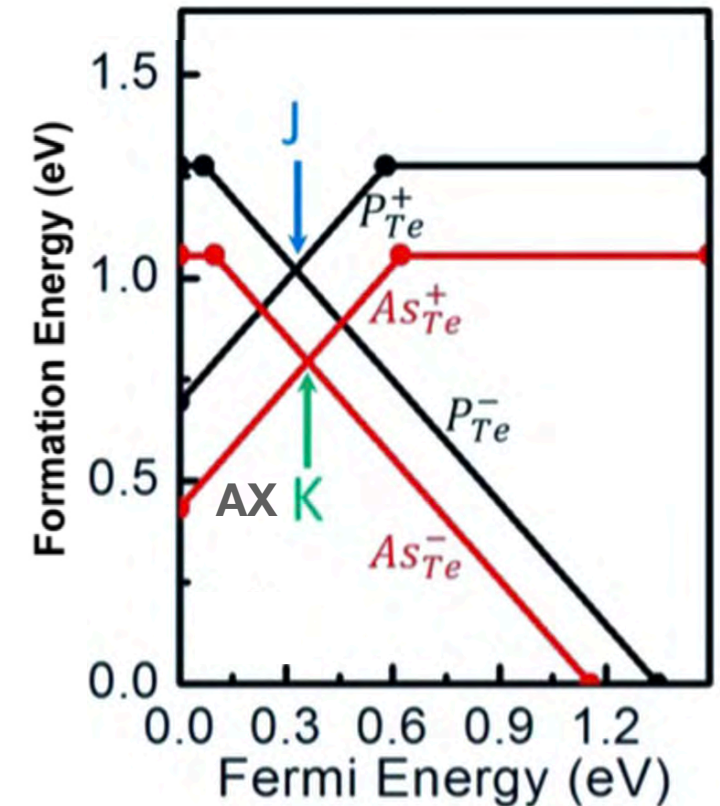
VBM

Wei and Zhang, Phys. Rev. B **66**, 155211 (2002)

These predictions indicate that it
Would be impossible to make
CdTe p-type with As or P

P and As are shallow acceptors
Self-compensation by AX centers
Fermi level pinned in the gap
negligible hole concentration

HSE06 hybrid functional
Supercell 64 atoms
no spin-orbit



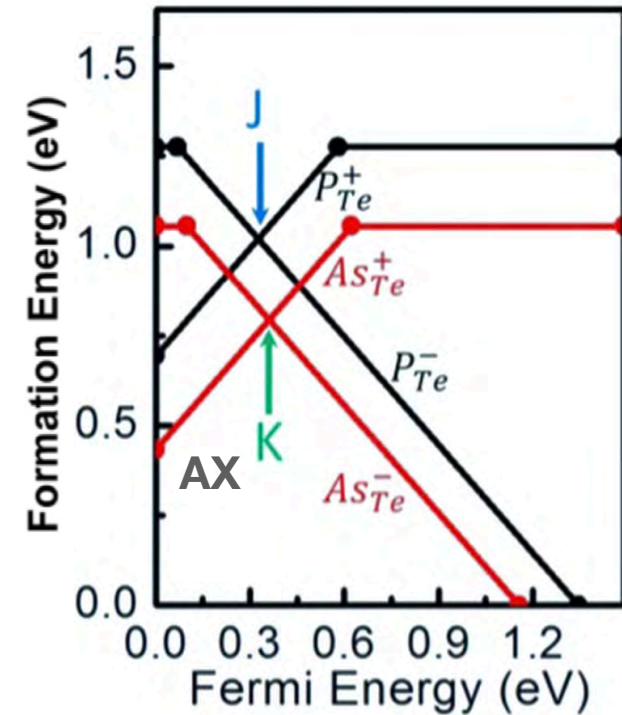
Yang *et. al.*, Semicond. Sci. Technol. **31**, 083002 (2016)

What do we know about acceptor impurities in CdTe from theory

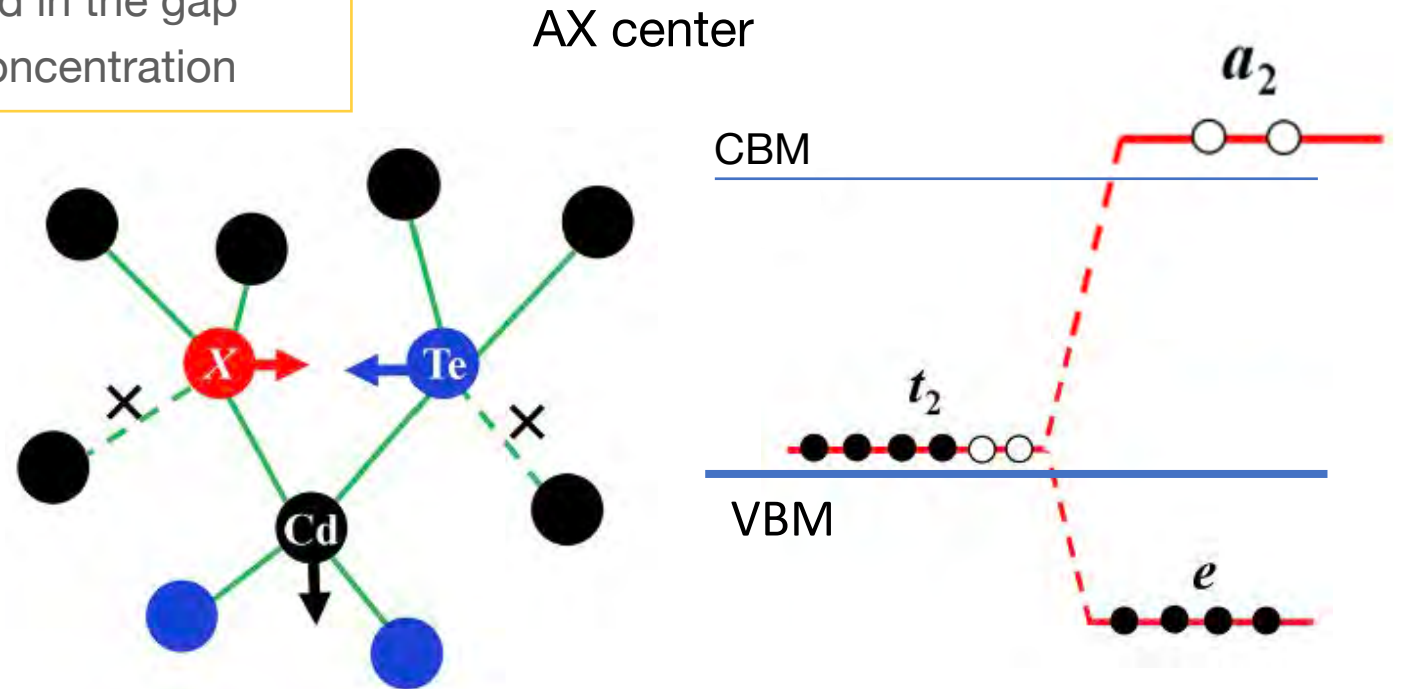
HSE06 hybrid functional
Supercell 64 atoms
no spin-orbit

P and As are shallow acceptors
Self-compensation by AX centers
Fermi level pinned in the gap
negligible hole concentration

Chadi, Phys. Rev. B **59**, 15181 (1999)



Yang *et. al.*, Semicond. Sci. Technol. **31**, 083002 (2016)

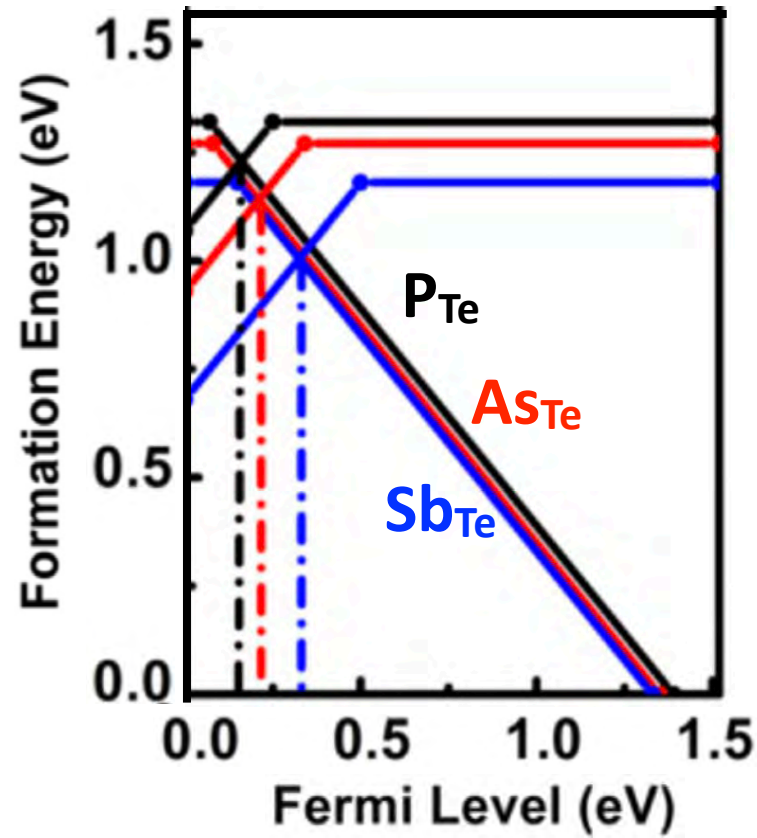


Large local lattice relaxation

Breaking two bonds and forming a X_{Te} -Te bond

What do we know about acceptor impurities in CdTe from theory

HSE06 hybrid functional
Supercell 216 atoms
no spin-orbit



(0/-) transition levels

P	70 meV
As	80 meV
Sb	150 meV

Fermi level pinned at the (+/-) level
→ negligible hole concentration
→ full self-compensation by AX centers

Had to use arguments based nonequilibrium
or kinetics to explain observed hole concentrations

**Te is a heavy atom → large splitting at the top of the valence band expected
due to spin-orbit coupling**

What are the effect of SOC?

band structure → push up the valence-band maximum (VBM)

defect levels

defect formation energies

Acceptor ionization energies and dependence on the supercell size

Stability of the AX centers

CdTe basic properties, different functionals

	PBEsol		HSE06		HSE(0.33)		Exp.
	no SOC	with SOC	no SOC	with SOC	no SOC	with SOC	
a (Å)	6.471		6.569		6.545		6.481
E_g (eV)	0.667	0.397	1.500	1.197	1.814	1.502	1.513 (T=300K)
ΔH_f (eV)	0.759		1.147		1.307		-0.96 -1.17

Sirdeshmukh *et al.*, Cryst. Res. Technol. **28**, 15 (1993)

Fonthal *et al.*, J. Phys. Chem. Solids **61**, 579 (2000)

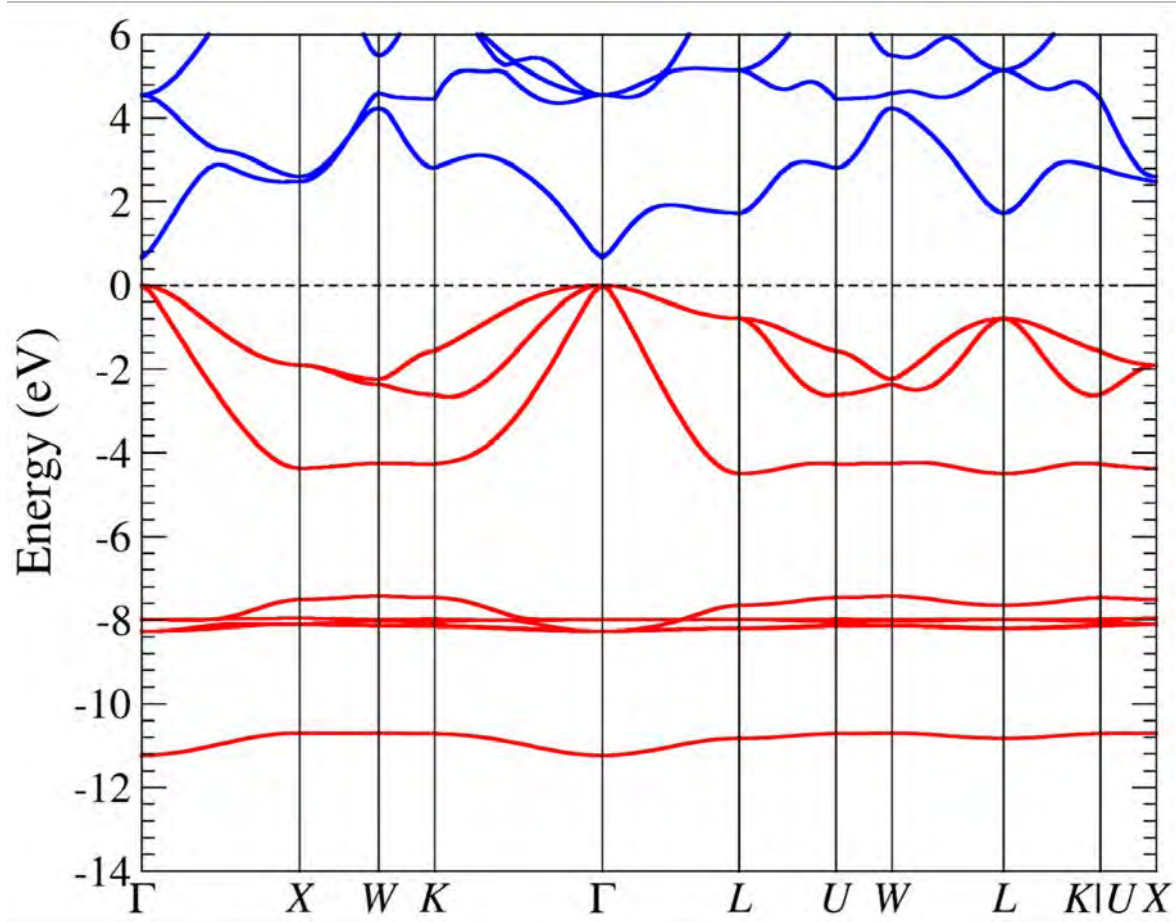
Dean, Lange's Handbook of Chemistry (McGraw-Hill, New York, 1999)

Yamaguchi *et al.*, Materials transactions, JIM **41**, 790 (2000)

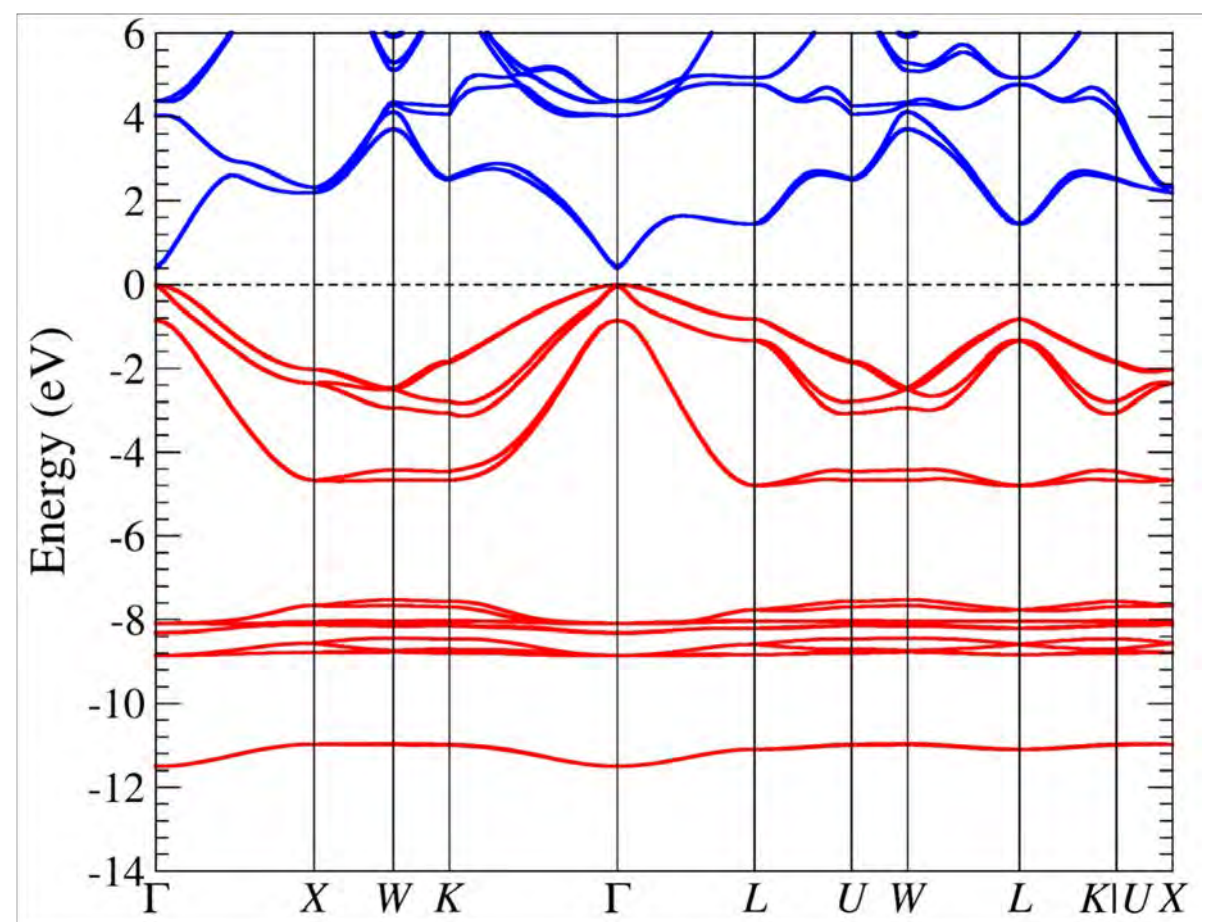
CdTe electronic structure, with and without SOC

DFT-GGA (PBEsol)

with SOC



Band gap = 0.667 eV

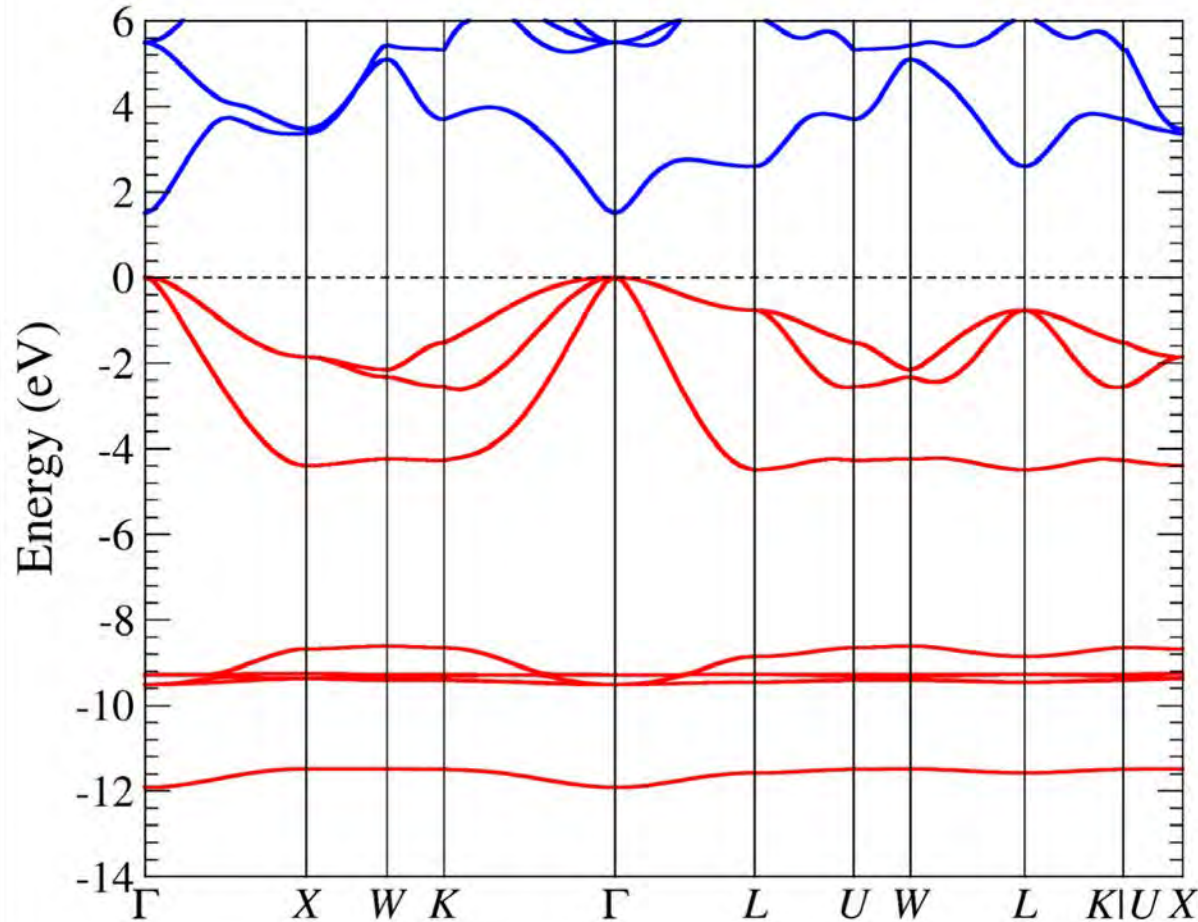


Band gap = 0.397 eV

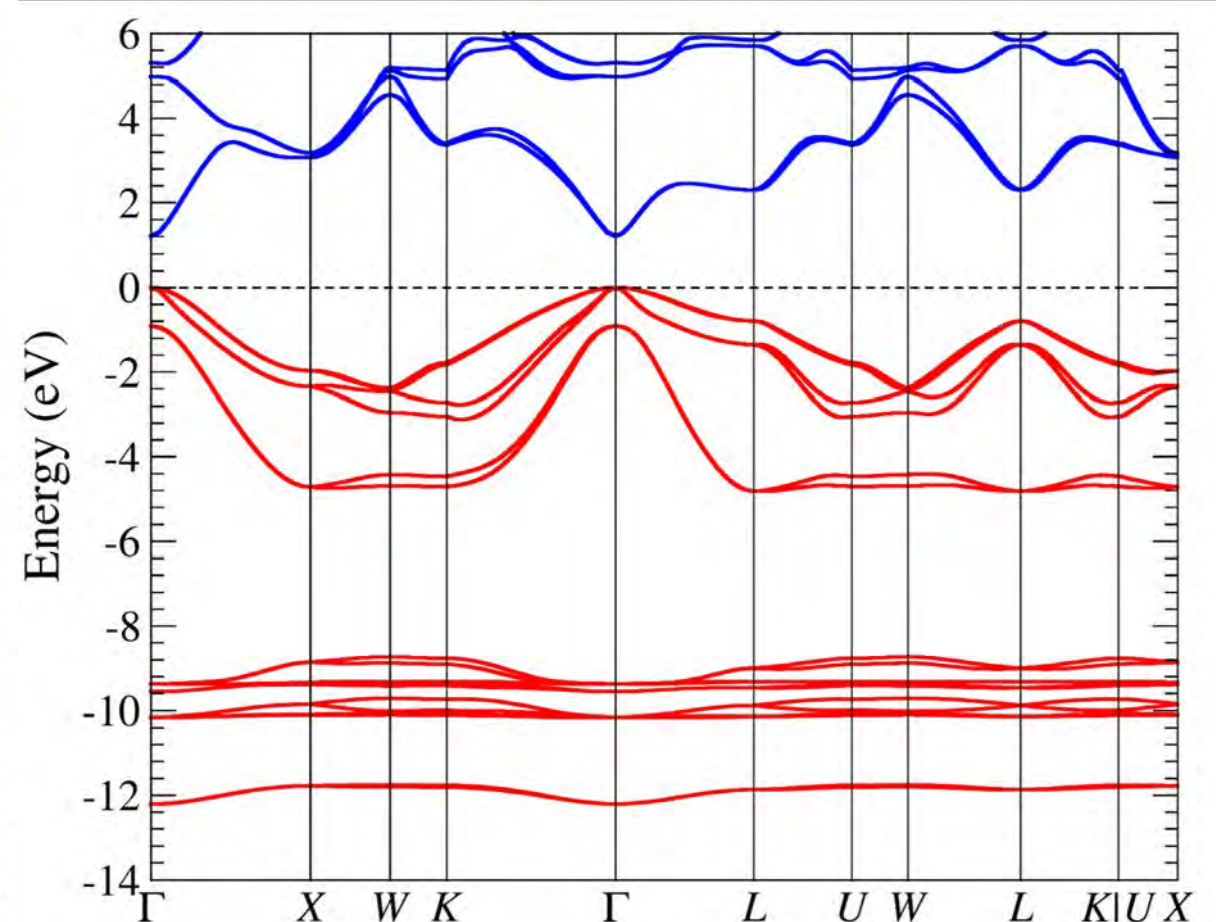
CdTe electronic structure, with and without SOC

HSE06

with SOC



Band gap = 1.500 eV

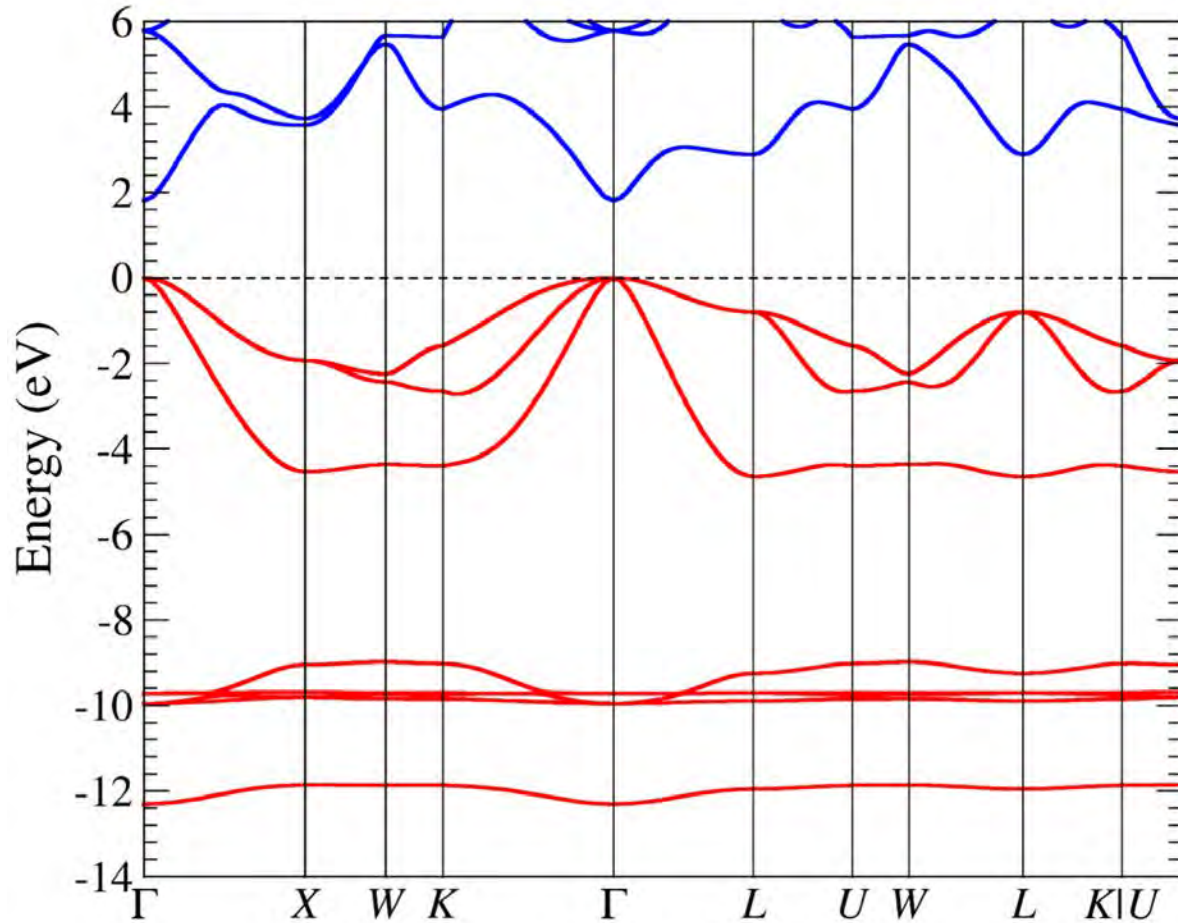


Band gap = 1.197 eV

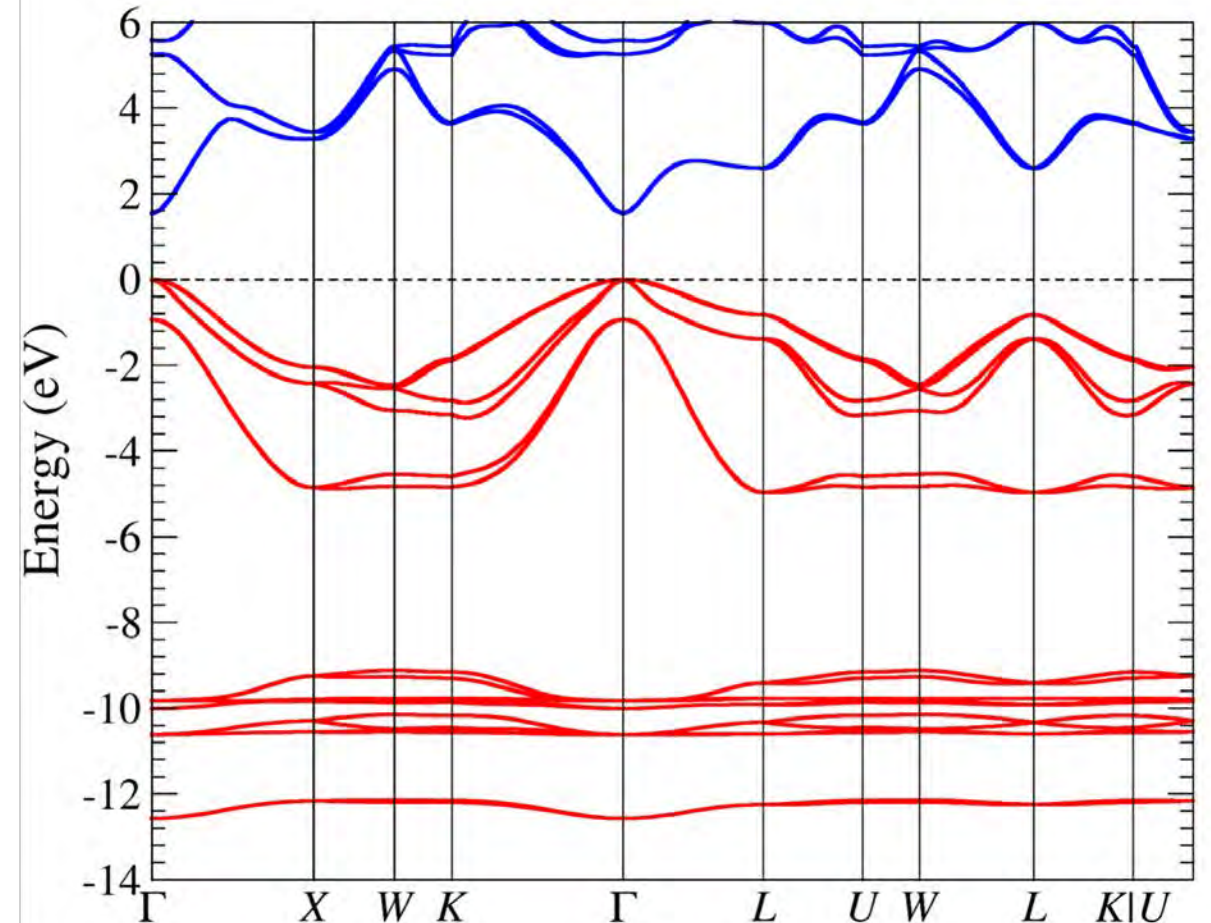
CdTe electronic structure, with and without SOC

HSE 33% mixing

with SOC



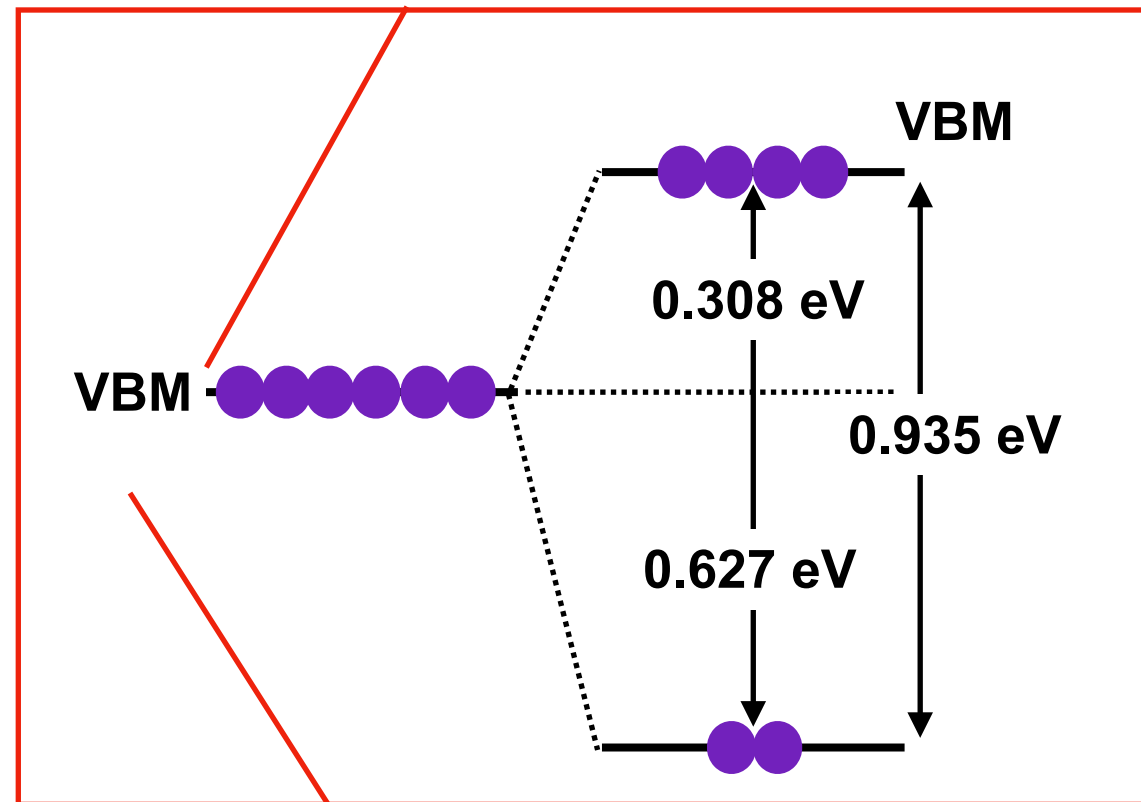
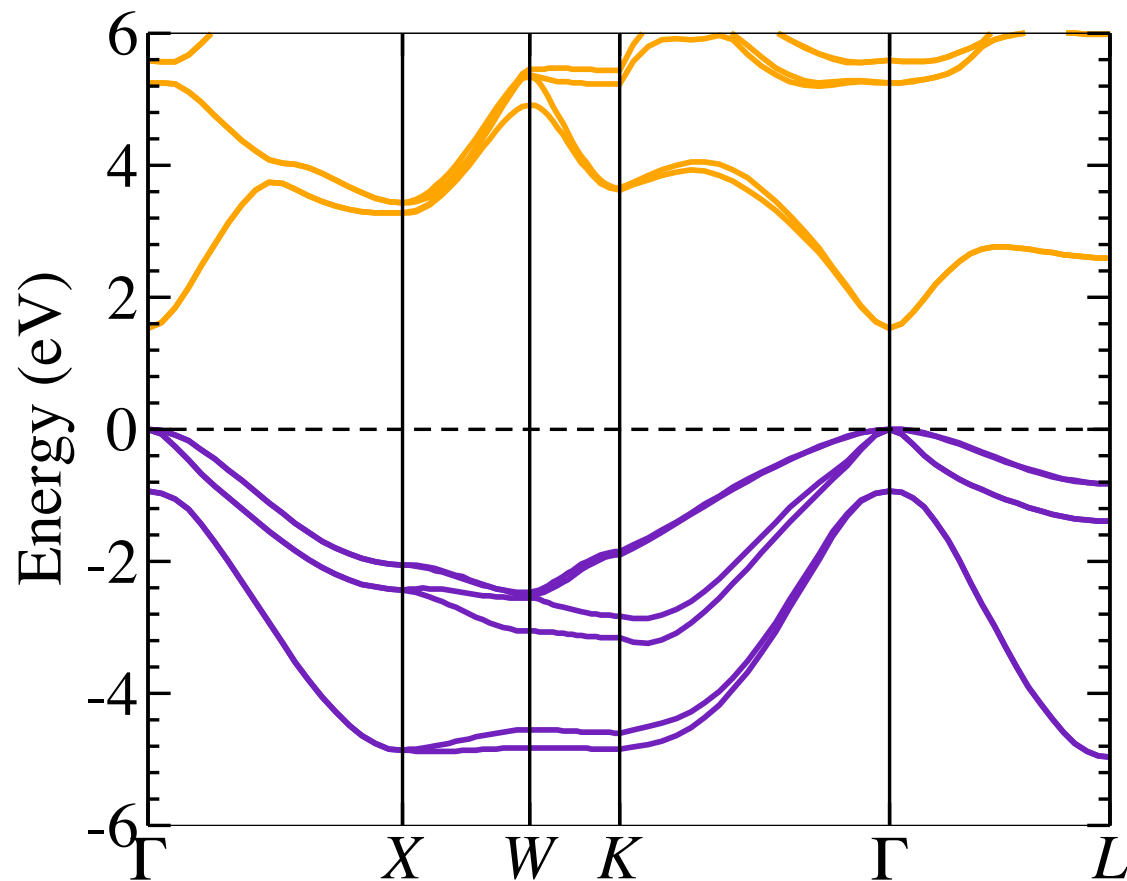
Band gap = 1.814 eV



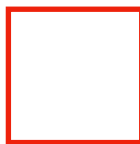
Band gap = 1.502 eV

CdTe electronic structure HSE $\alpha = 0.33$

spin-orbit splitting



see also Pan *et al.*, Phys. Rev. B **98** 054108 (2018)



CdTe effective masses

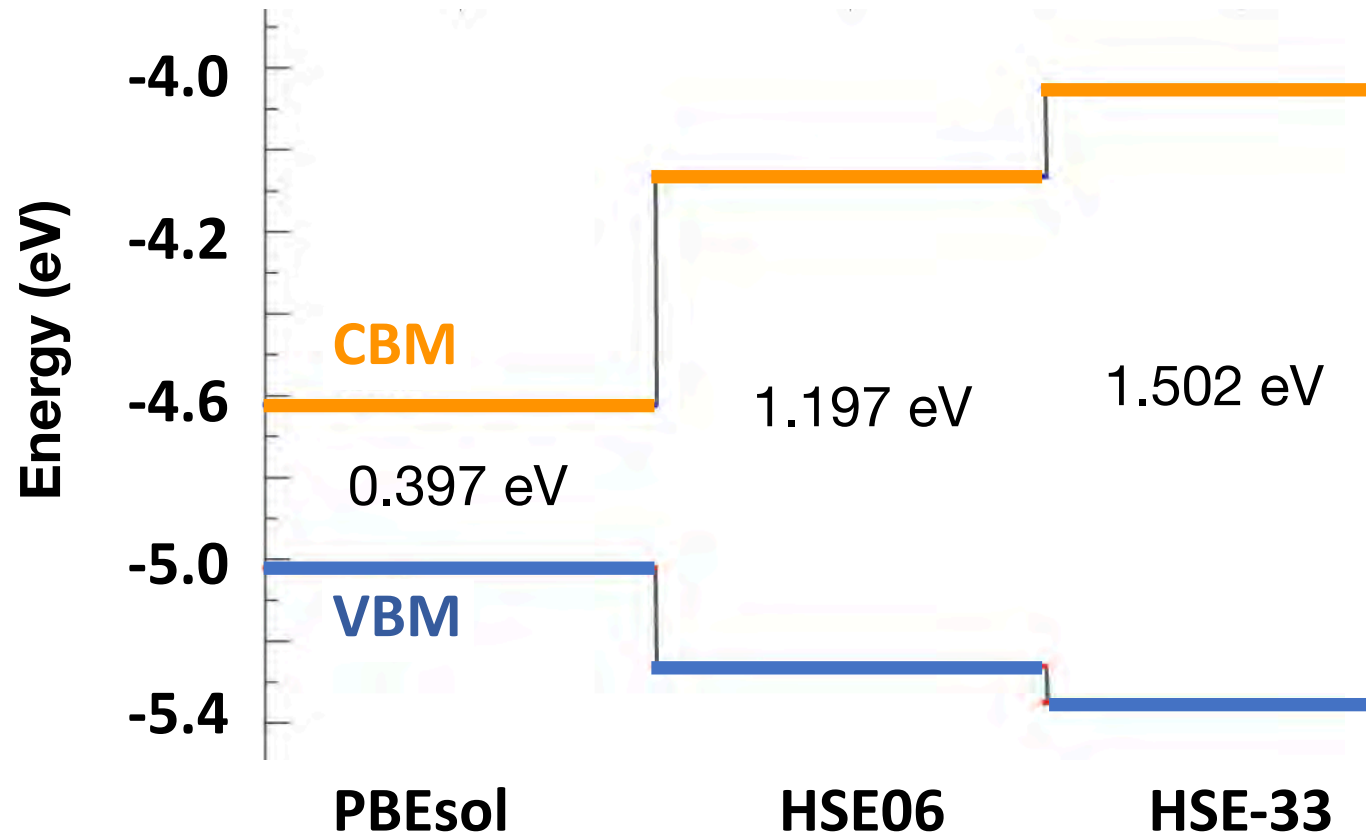
HSE $\alpha = 0.33$

	PBEsol		HSE06		HSE(0.33)		Exp.
Effective mass	no SOC	with SOC	no SOC	with SOC	no SOC	with SOC	
[100] G-X	$m_{hh} = 0.504$ $m_{lh} = 0.064$ $m_e = 0.063$	$m_{hh} = 0.440$ $m_{lh} = 0.060$ $m_{sp} = 0.246$ $m_e = 0.053$	$m_{hh} = 0.533$ $m_{lh} = 0.103$ $m_e = 0.100$	$m_{hh} = 0.474$ $m_{lh} = 0.113$ $m_{sp} = 0.307$ $m_e = 0.094$	$m_{hh} = 0.522$ $m_{lh} = 0.113$ $m_e = 0.110$	$m_{hh} = 0.471$ $m_{lh} = 0.129$ $m_{sp} = 0.321$ $m_e = 0.106$	$m_{hh} = 0.72$ $m_{lh} = 0.1$ $m_e = 0.094$
[110] G-K	$m_{hh} = 2.586, 0.502$ $m_{lh} = 0.060$ $m_e = 0.065$	$m_{hh} = 0.797$ $m_{lh} = 0.060$ $m_{sp} = 0.243$ $m_e = 0.055$	$m_{hh} = 2.581, 0.530$ $m_{lh} = 0.090$ $m_e = 0.102$	$m_{hh} = 0.865$ $m_{lh} = 0.105$ $m_{sp} = 0.303$ $m_e = 0.096$	$m_{hh} = 2.612, 0.520$ $m_{lh} = 0.097$ $m_e = 0.111$	$m_{hh} = 0.860$ $m_{lh} = 0.117$ $m_{sp} = 0.317$ $m_e = 0.108$	$m_{hh} = 0.81$ $m_{lh} = 0.12$ $m_e = 0.096$
[111] G-L	$m_{hh} = 1.083$ $m_{lh} = 0.056$ $m_e = 0.062$	$m_{hh} = 1.112$ $m_{lh} = 0.055$ $m_{sp} = 0.244$ $m_e = 0.051$	$m_{hh} = 1.132$ $m_{lh} = 0.085$ $m_e = 0.100$	$m_{hh} = 1.024$ $m_{lh} = 0.101$ $m_{sp} = 0.304$ $m_e = 0.094$	$m_{hh} = 1.127$ $m_{lh} = 0.092$ $m_e = 0.110$	$m_{hh} = 1.191$ $m_{lh} = 0.111$ $m_{sp} = 0.320$ $m_e = 0.106$	$m_e = 0.095$
ionization potential (eV)	-5.022		-5.261		-5.350		-5.6, -5.8

HSE-33% improves the description of effective masses and ionization potential

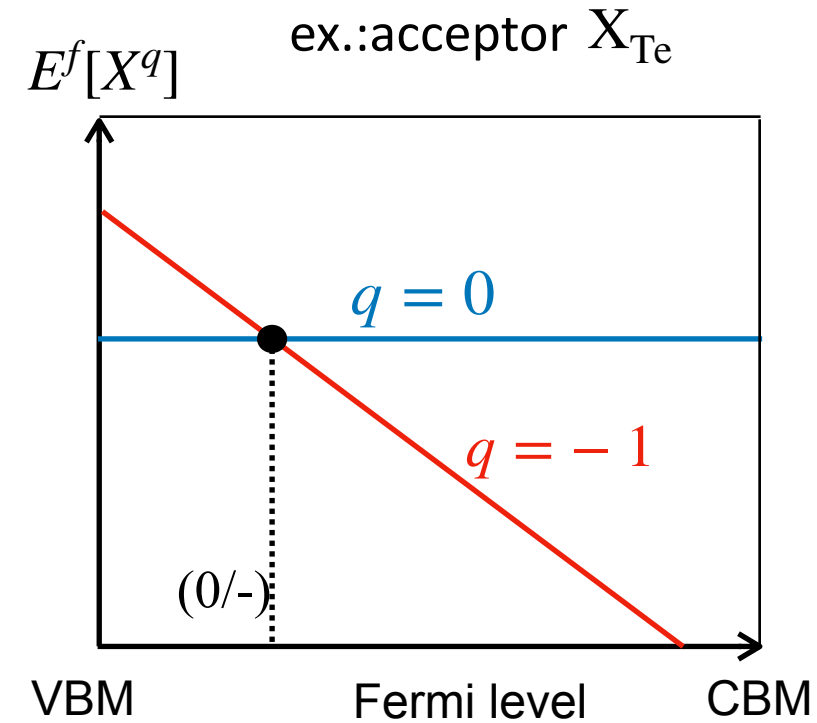
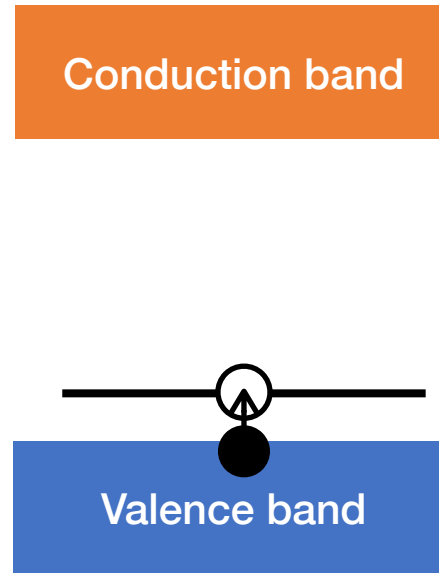
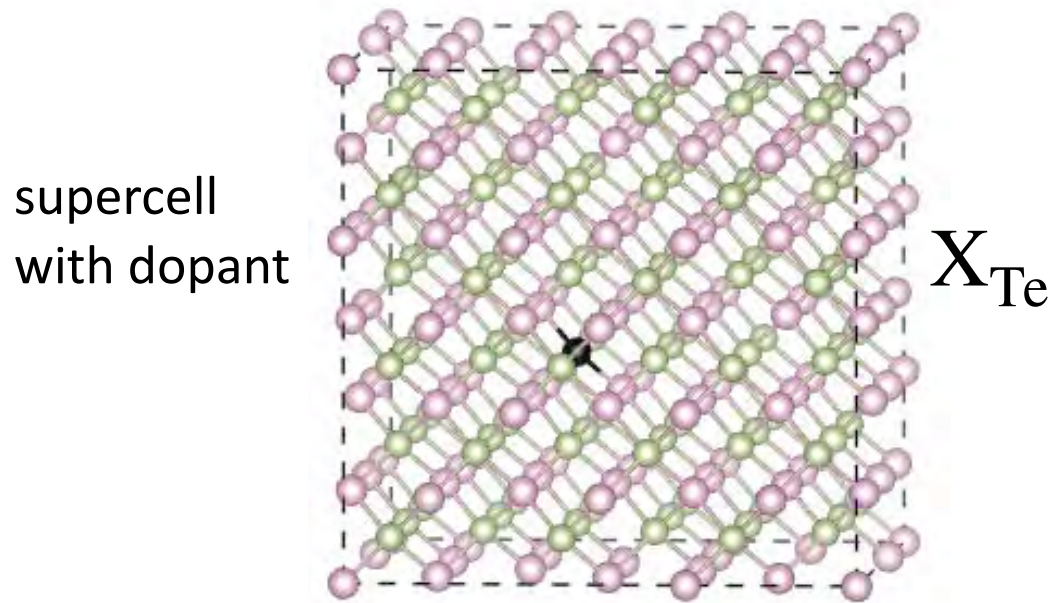
CdTe ionization potential

different functionals



affects both valence and conduction band, almost equally

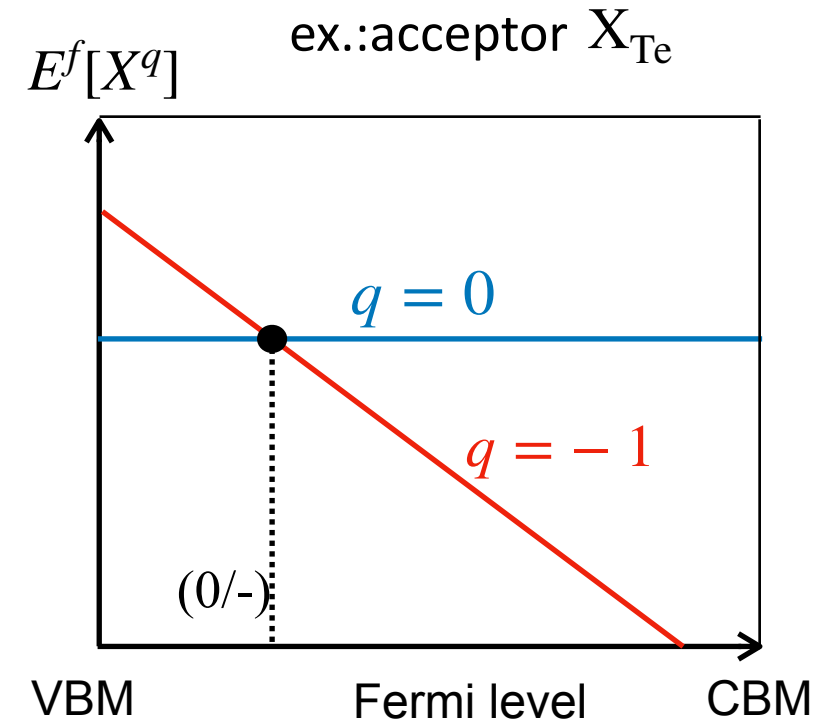
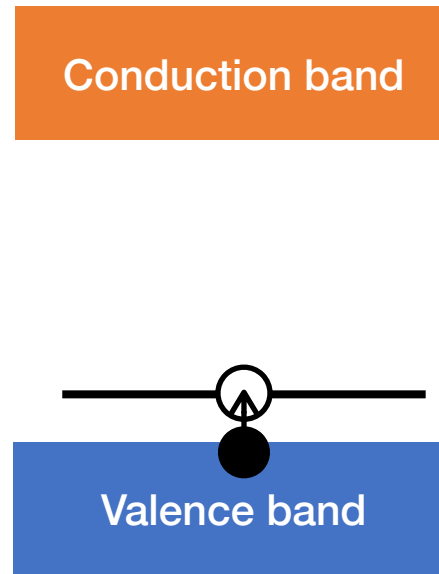
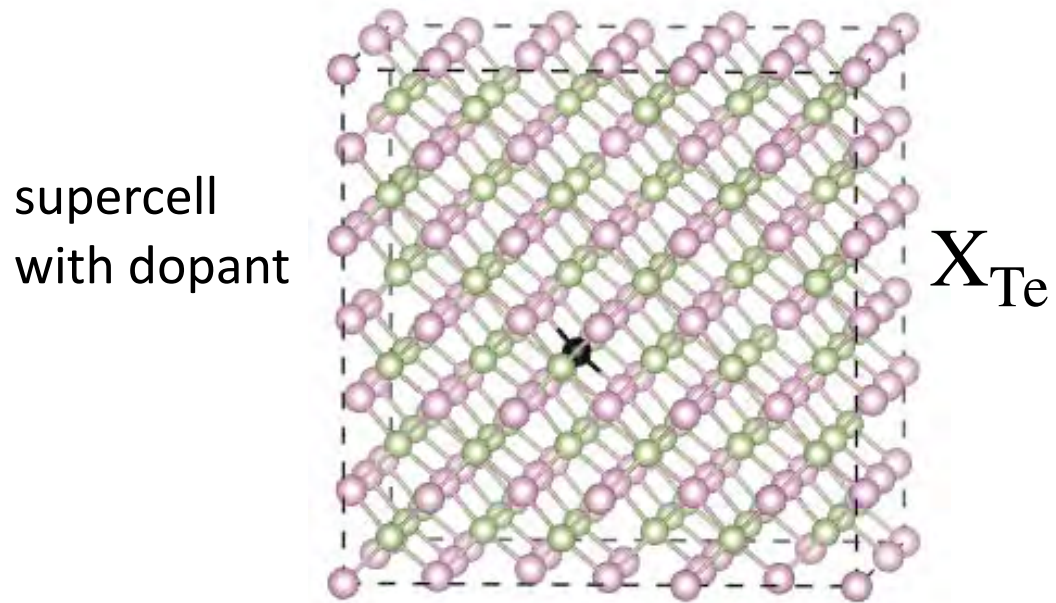
Dopant/defect formation energy and transition level



$$E^f[X^q] = E_{tot}[X^q] - E_{tot}[bulk] + \sum_i n_i \mu_i + q(\epsilon_f + E_{VBM}) + \Delta^q$$

Freysoldt *et. al.*, Rev. Mod. Phys. **86**, 253 (2014)

Dopant/defect formation energy and transition level



$$E^f[X^q] = E_{tot}[X^q] - E_{tot}[bulk] + \sum_i n_i \mu_i + q(\epsilon_f + E_{VBM}) + \Delta^q$$

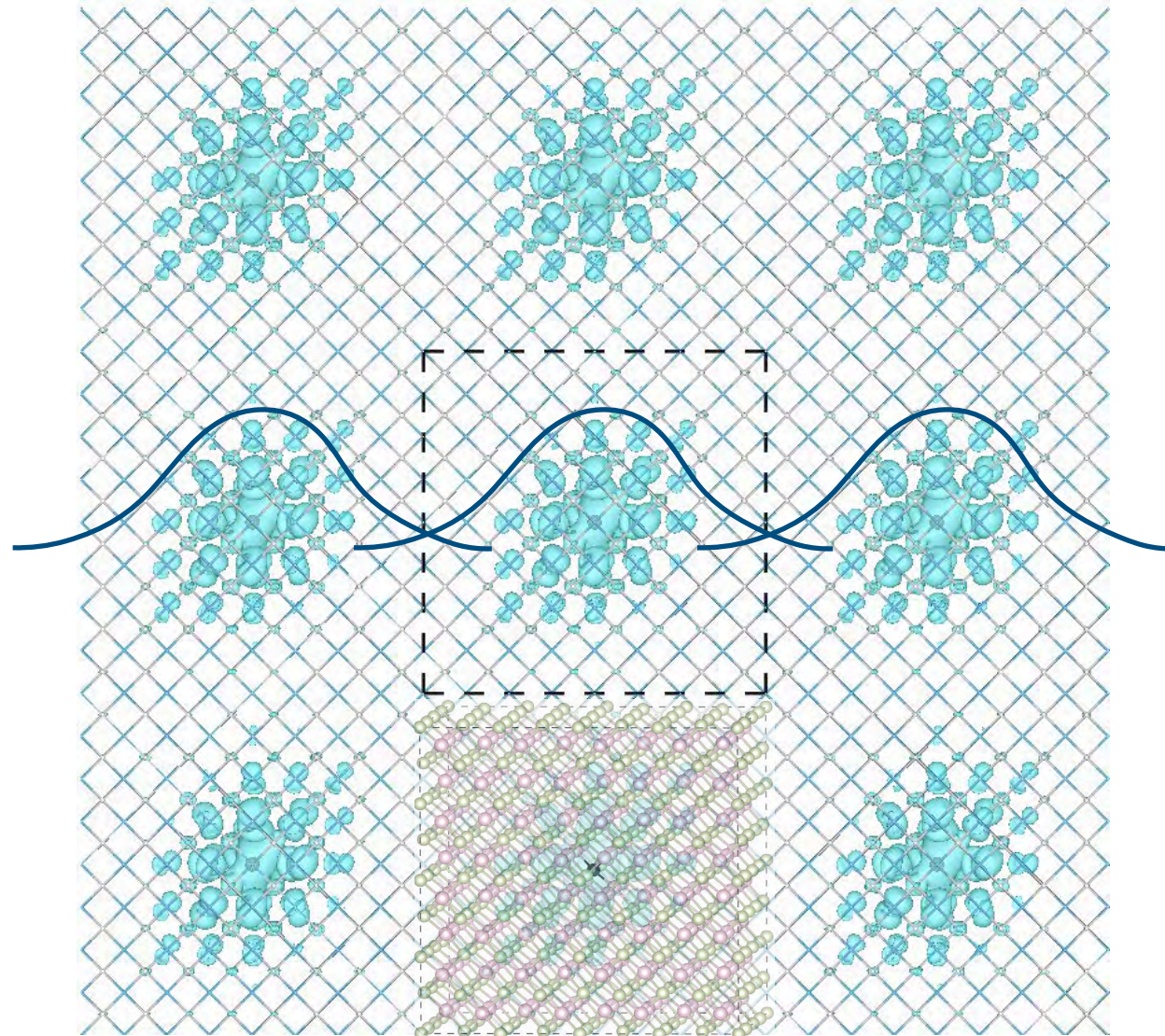
Typical supercell sizes are not large enough
to describe an isolated shallow center

Swift *et al.*, Npj Comput. Mater. **6**, 181 (2020)

King and Wang, Phys. Rev. Appl. **18**, 064001 (2022)

Freysoldt *et. al.*, Rev. Mod. Phys. **86**, 253 (2014)

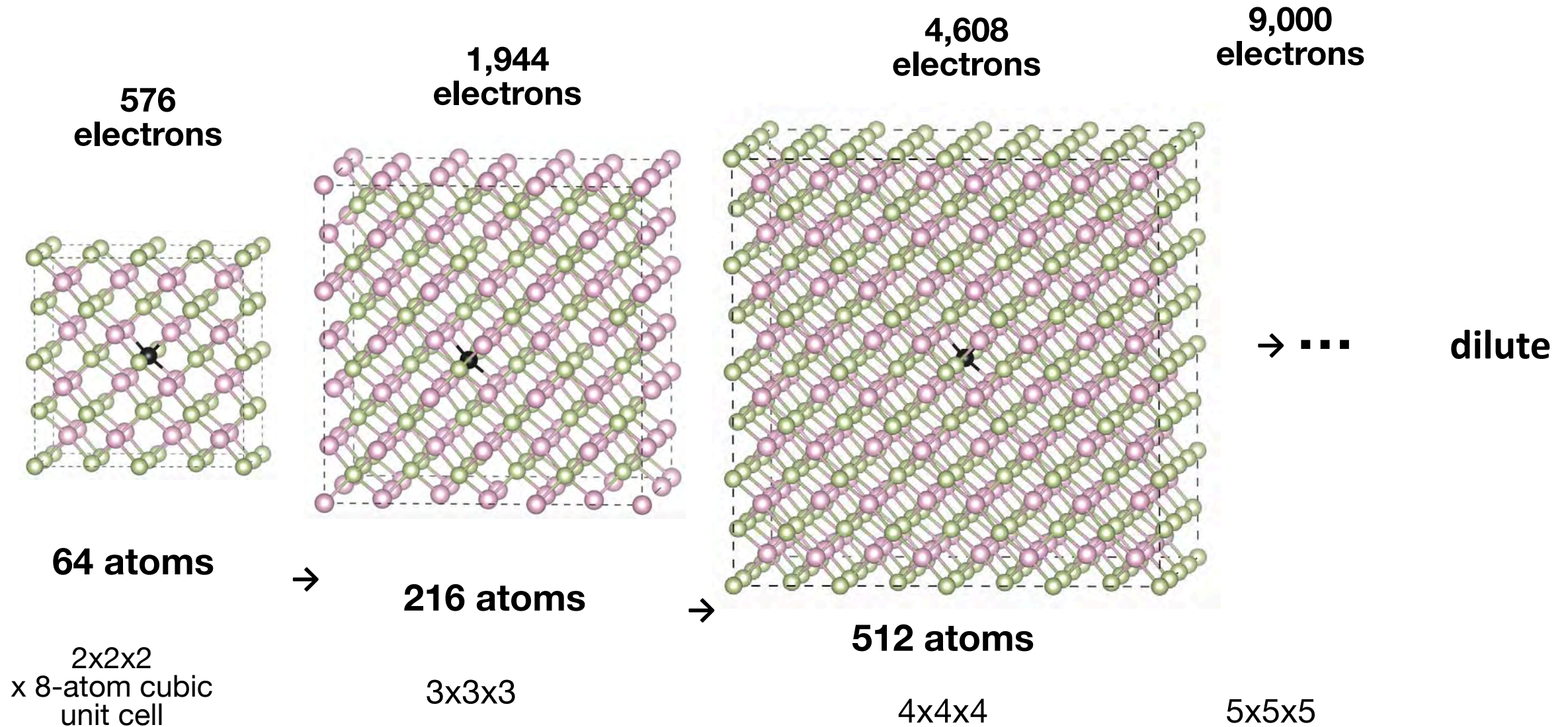
Interactions between defects in neighboring cells



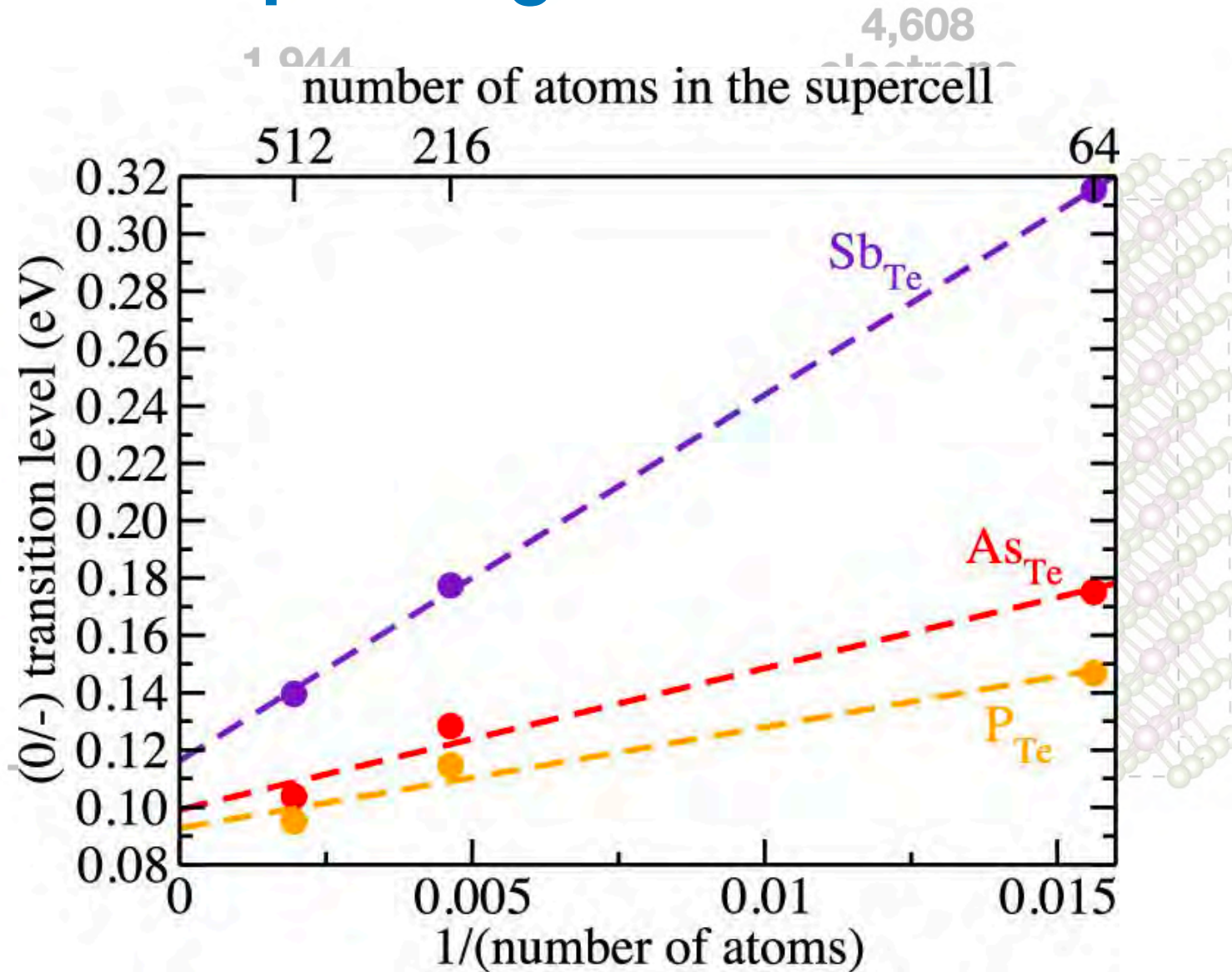
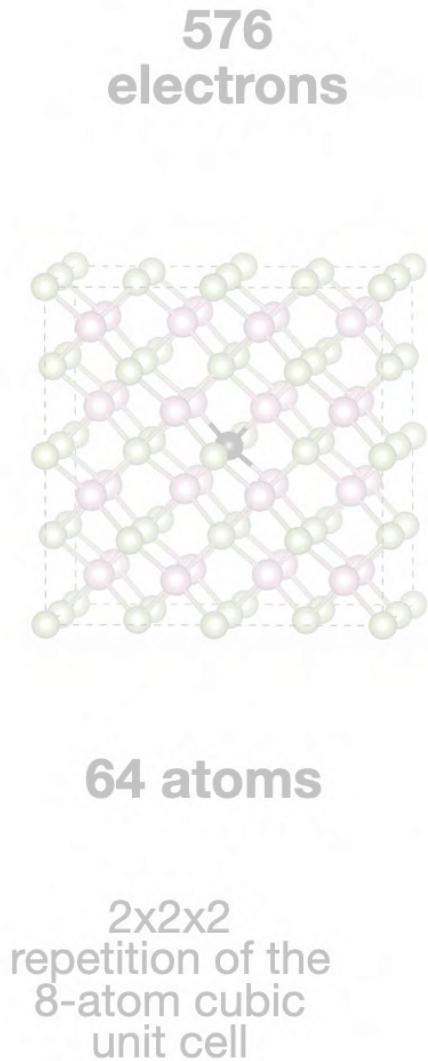
For typical supercell sizes of few 100 atoms

Errors in transition levels (~ 0.1 eV)
are of the order of the transition-level values
for shallow centers

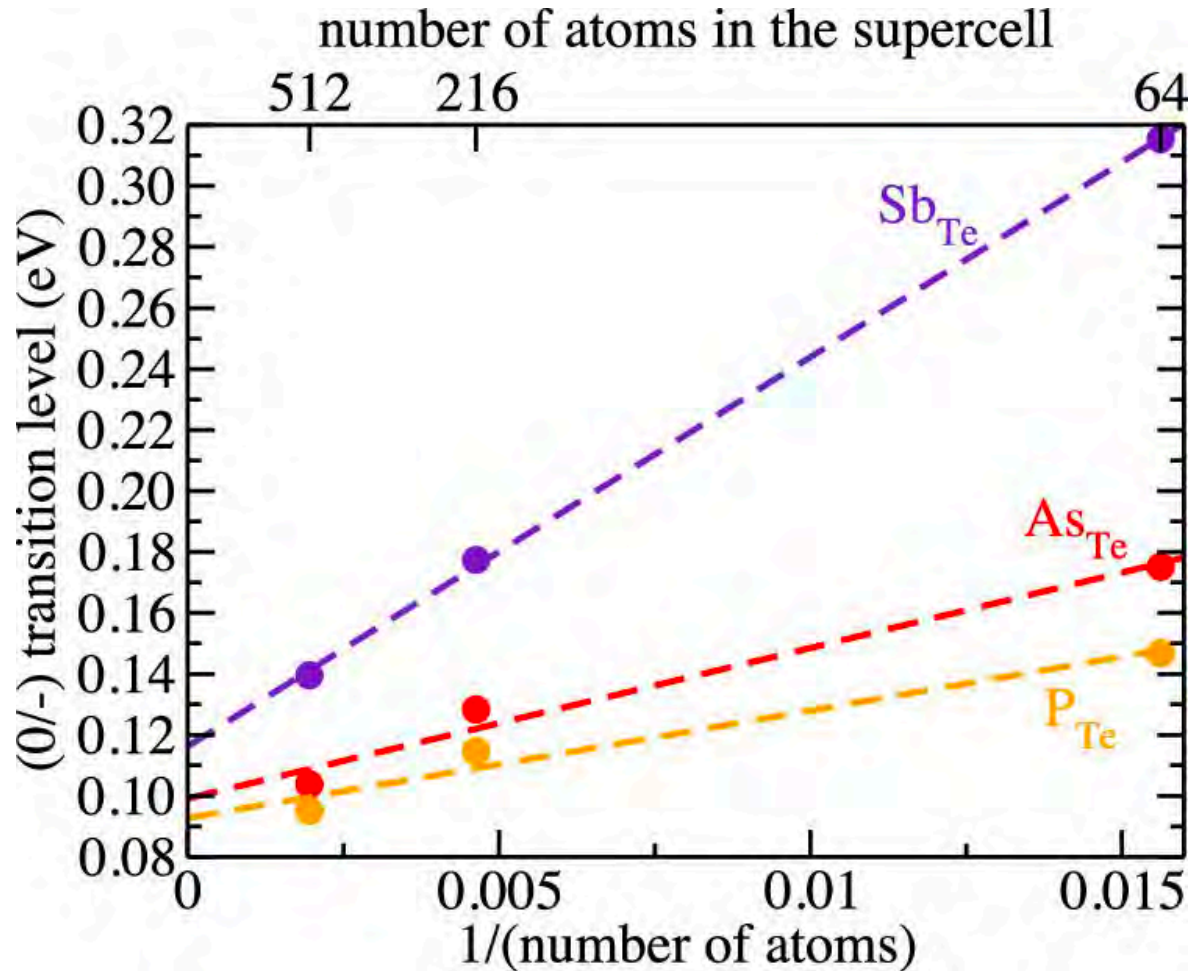
Extrapolating to the dilute limit



Extrapolating to the dilute limit



Extrapolating to the dilute limit



At dilute limit:

P(0/-) = 93 meV

As(0/-) = 99 meV

Sb(0/-) = 116 meV

Exp.

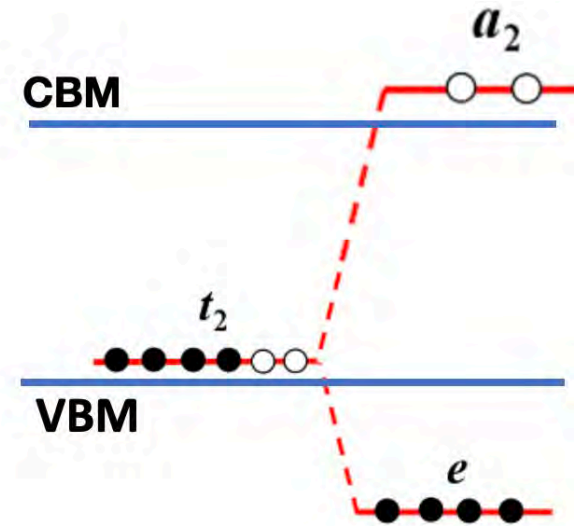
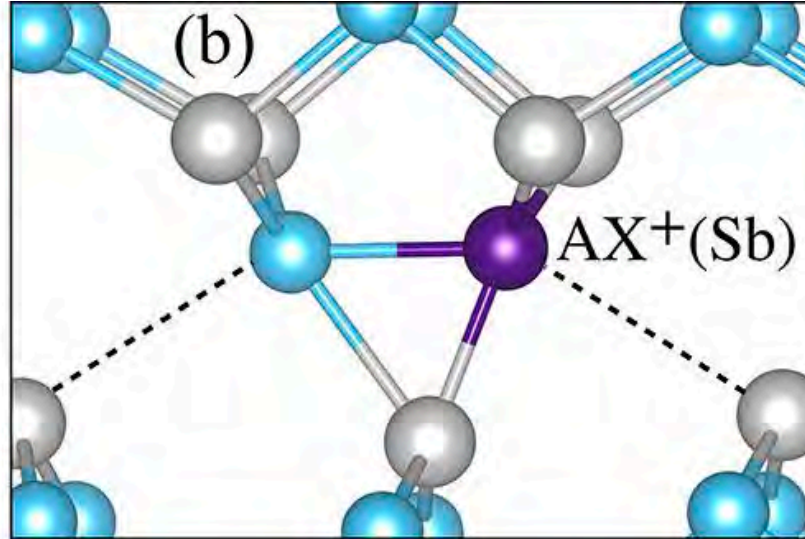
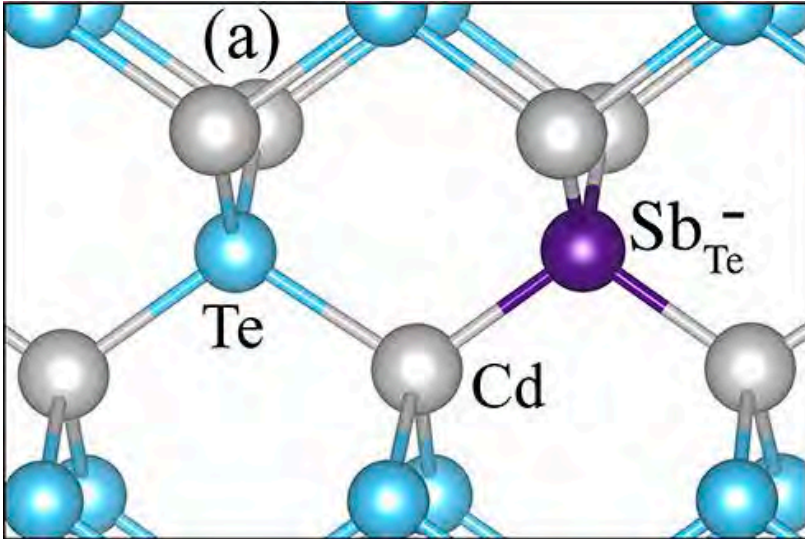
P(0/-) = 87 meV

As(0/-) = 94 meV

Sb(0/-) = 103 meV

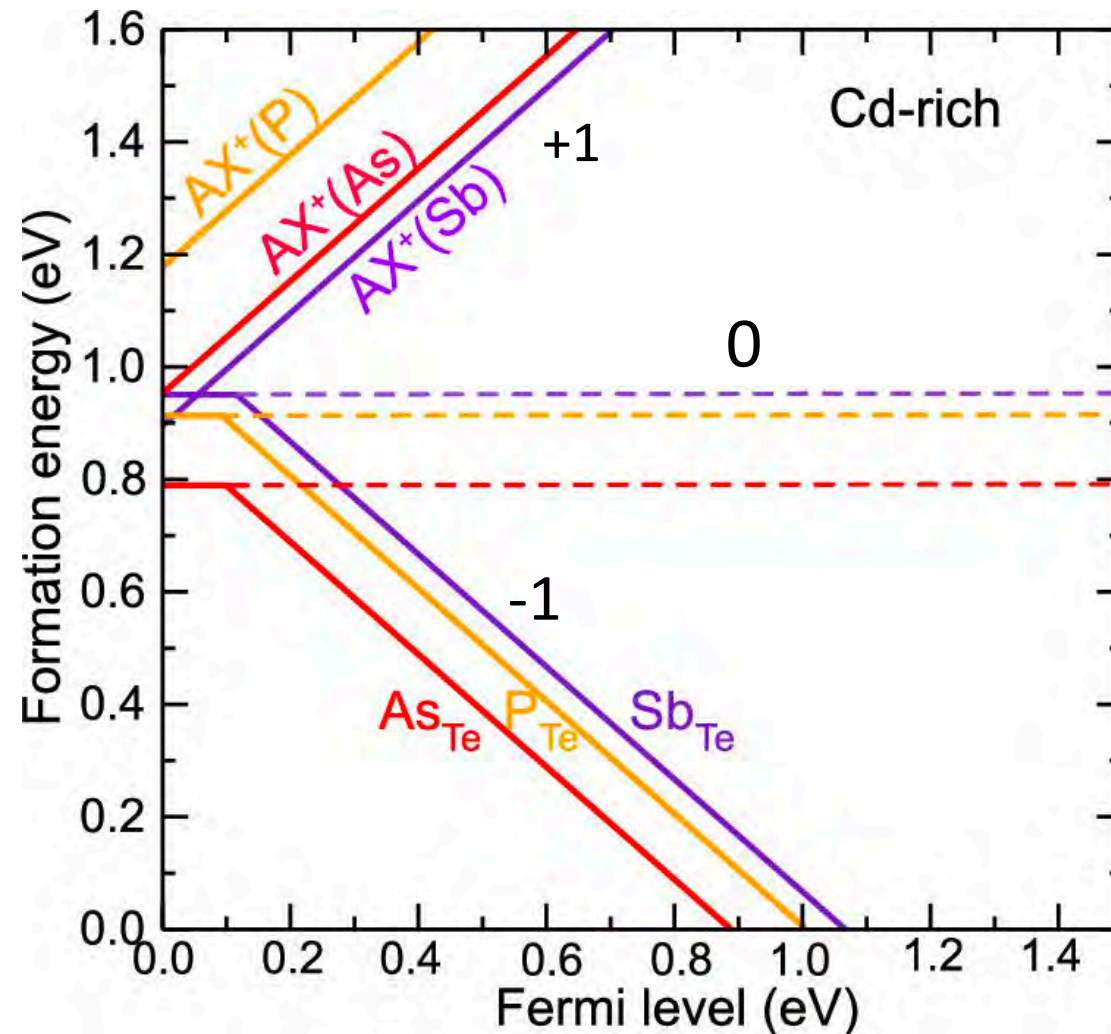
Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Substitutional acceptors vs. AX centers



electronic energy gain
vs.
lattice strain loss

Substitutional acceptors vs. AX centers



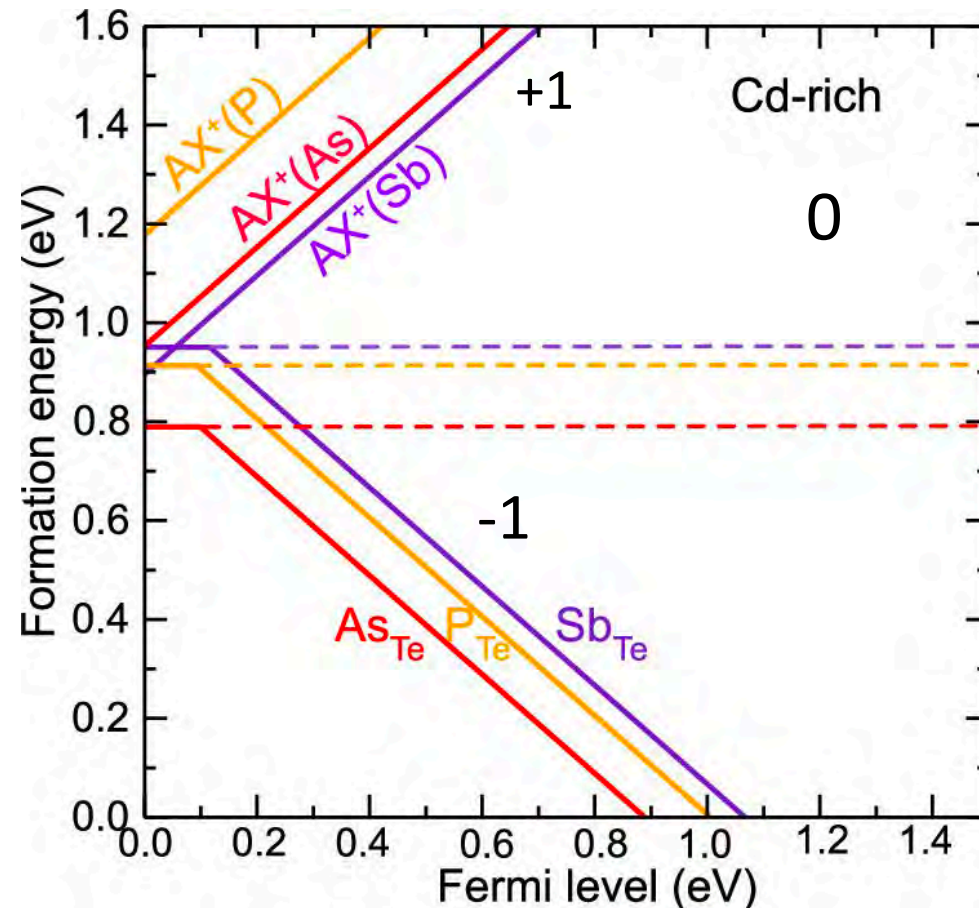
- AX centers are not stable
- P, As, and Te are shallow donors
- (0/-) ionization energies close to expected from hydrogen model

$$13.6 \text{ eV } m^*/\epsilon^2 \approx 100 \text{ meV}$$

Chatratin *et al.*, J. Phys. Chem. Lett. **14**, 273 (2023)

Substitutional acceptors vs. AX centers

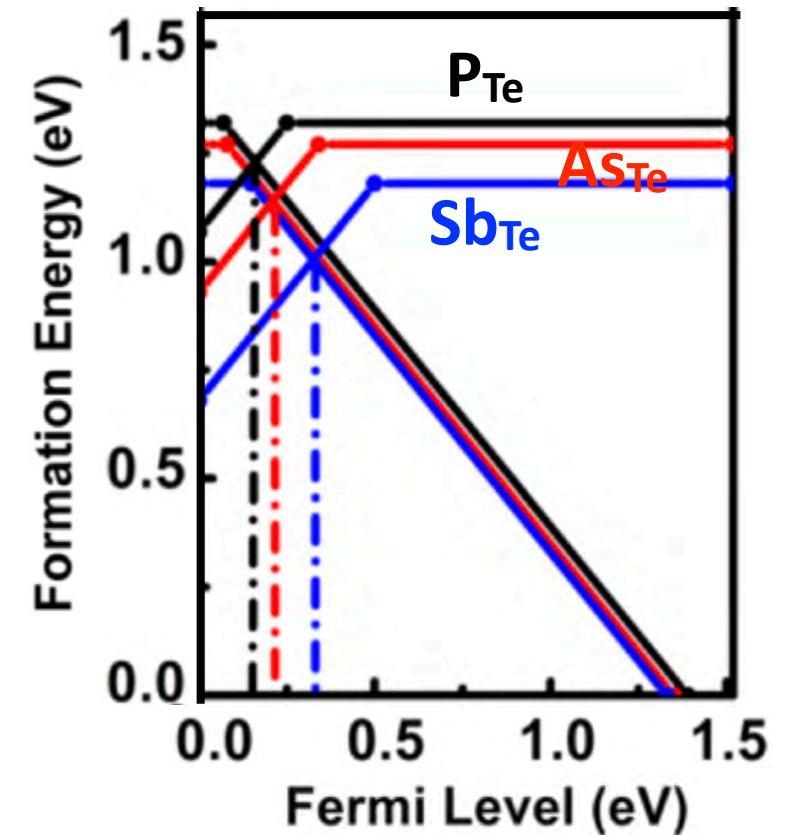
HSE-33 hybrid functional
with spin-orbit
Extrapolated to the dilute limit



Chatratin et al., J. Phys. Chem. Lett. **14**, 273 (2023)

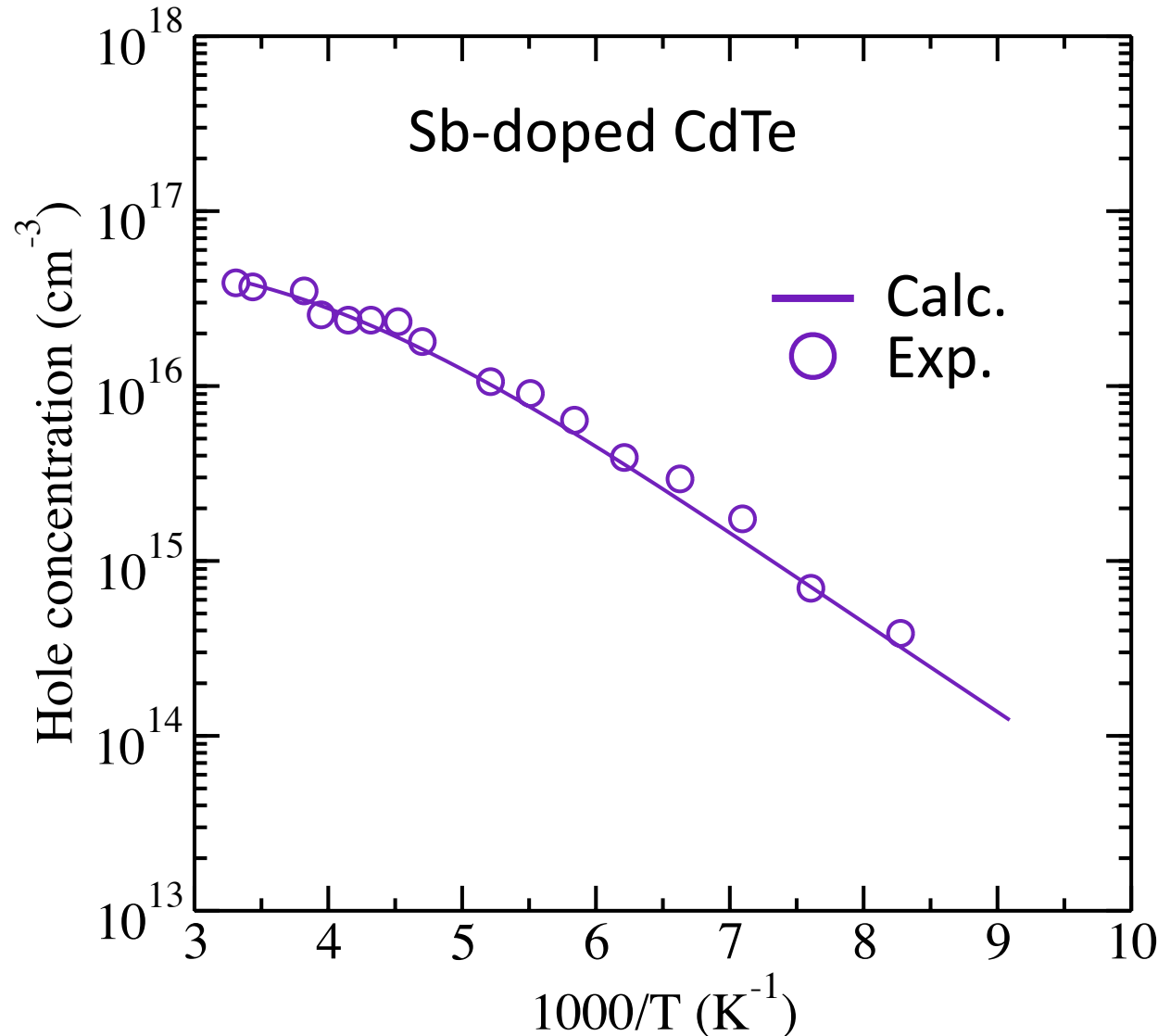
31

HSE06 hybrid functional
no spin-orbit
Supercell 216 atoms



Dou *et. al.*, Phys. Rev. Appl. **15**, 054045 (2021)

Calculated hole concentration for Sb-doped CdTe



Use calc. $E_a = 116$ meV

Fit: $[\text{Sb}] = [N_a] = 0.58 \times 10^{17} \text{ cm}^{-3}$

$$\frac{p(p + N_d)}{N_a - N_d - p} = \frac{N_V}{\beta} e^{-E_a/k_B T}$$

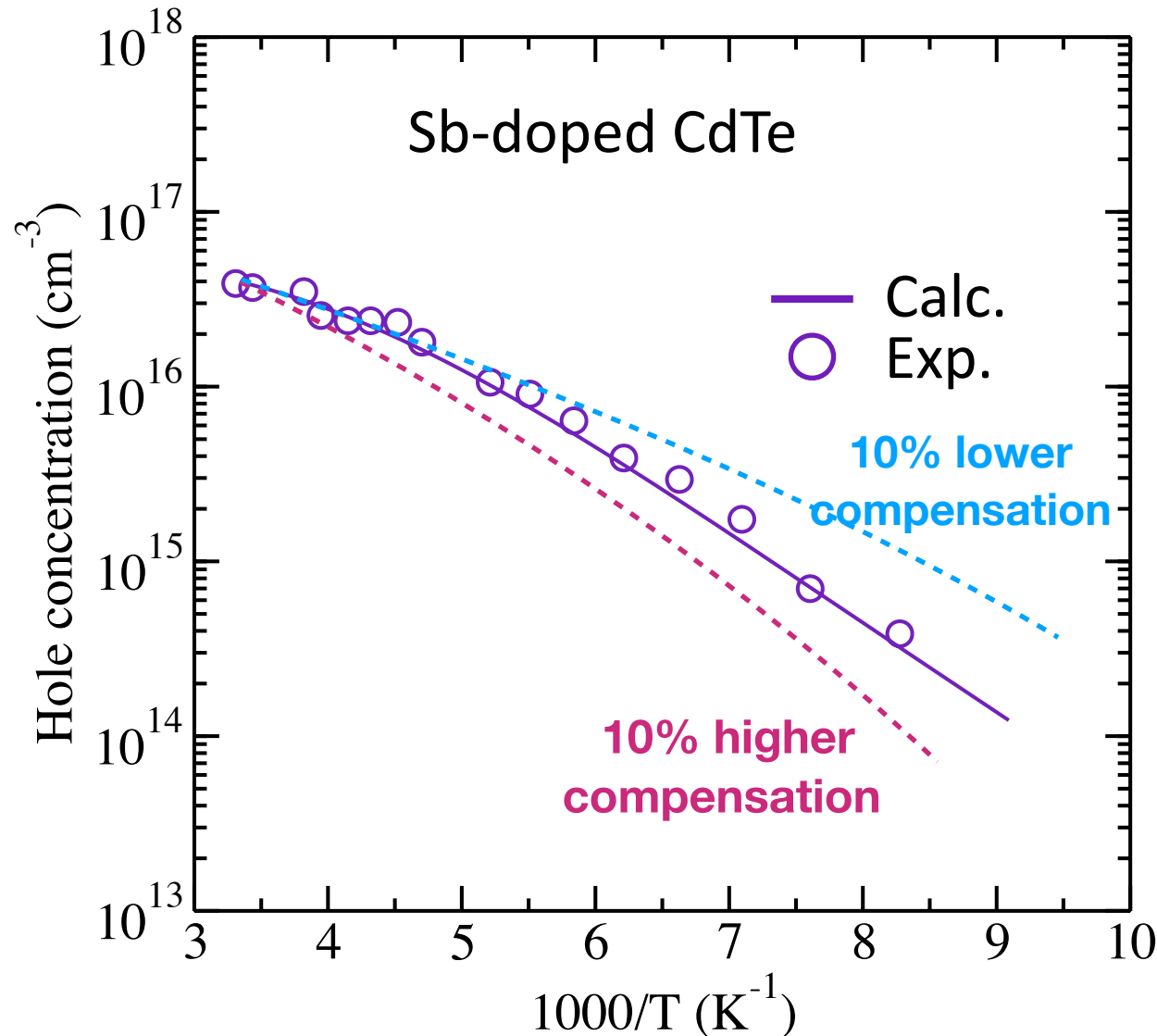
Blakemore, Semiconductor statistics
(Courier Corp., 2002)

N_d are AX centers $\sim 3\%$ of $[\text{Sb}]$

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for Sb-doped CdTe

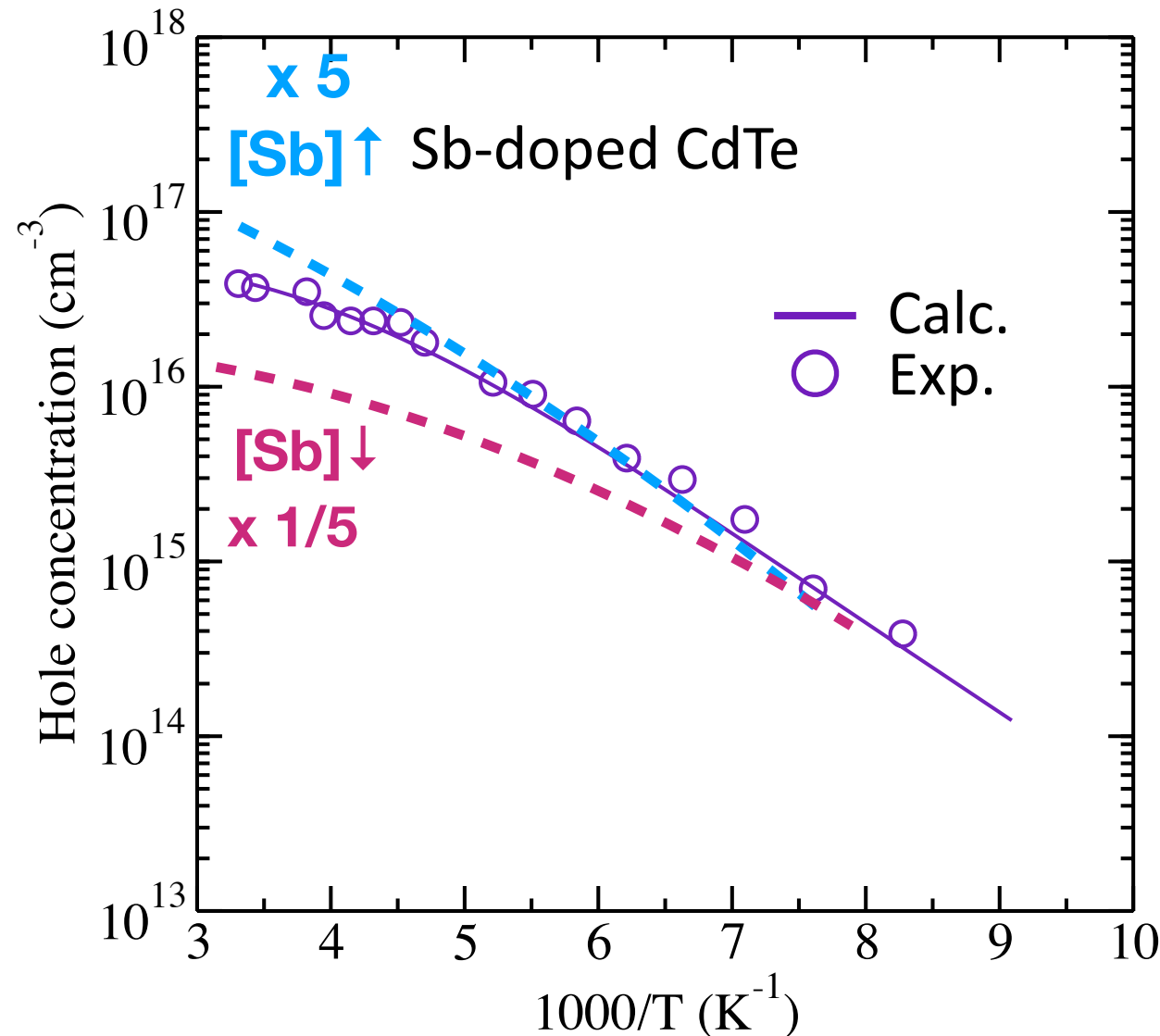


Changing the level of compensation (assuming unknown donor) makes the agreement with experiments worse at low temperatures

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for Sb-doped CdTe

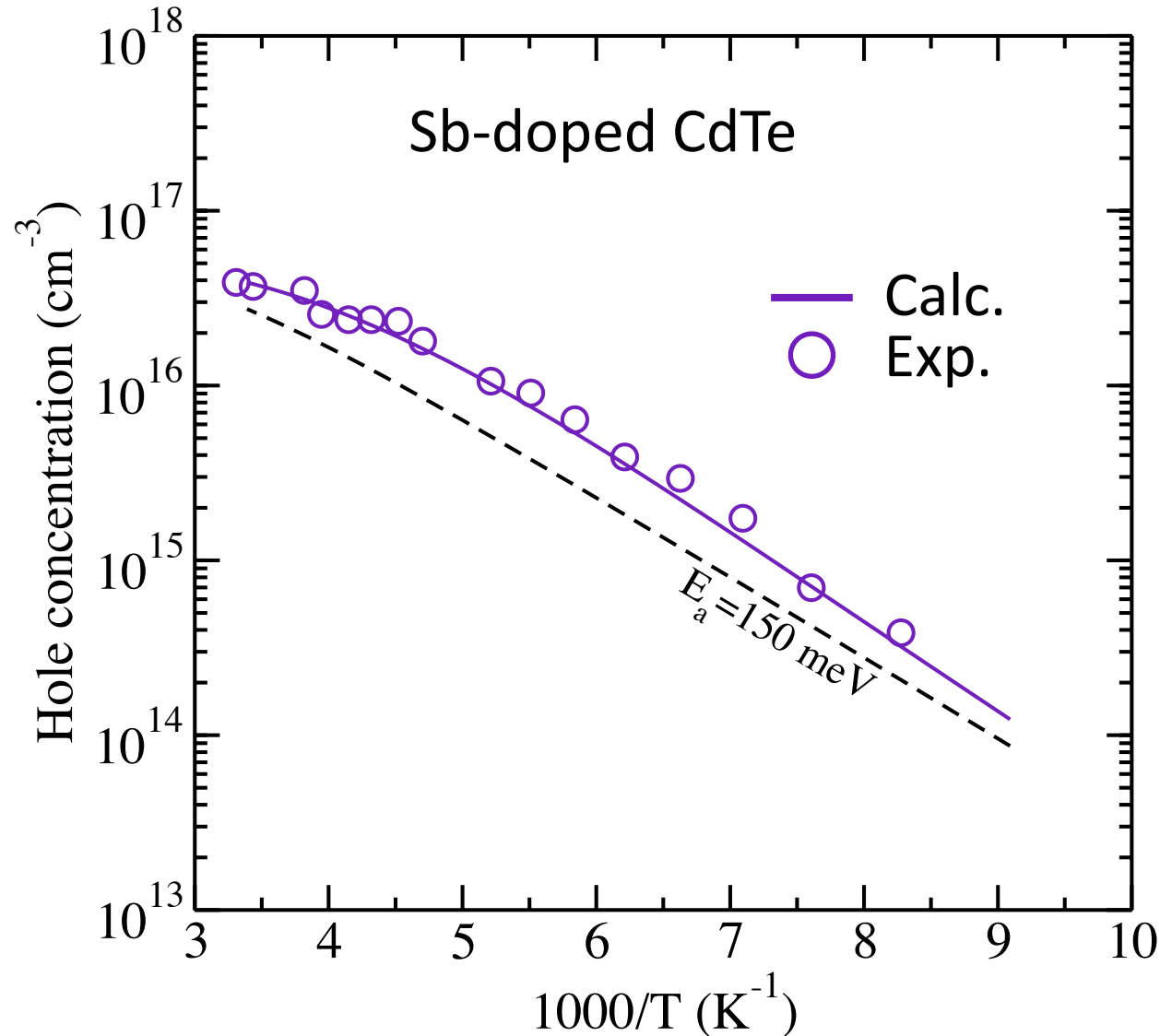


Changing the total Sb concentration by a factor of 5 makes the agreement worse at room temperature

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

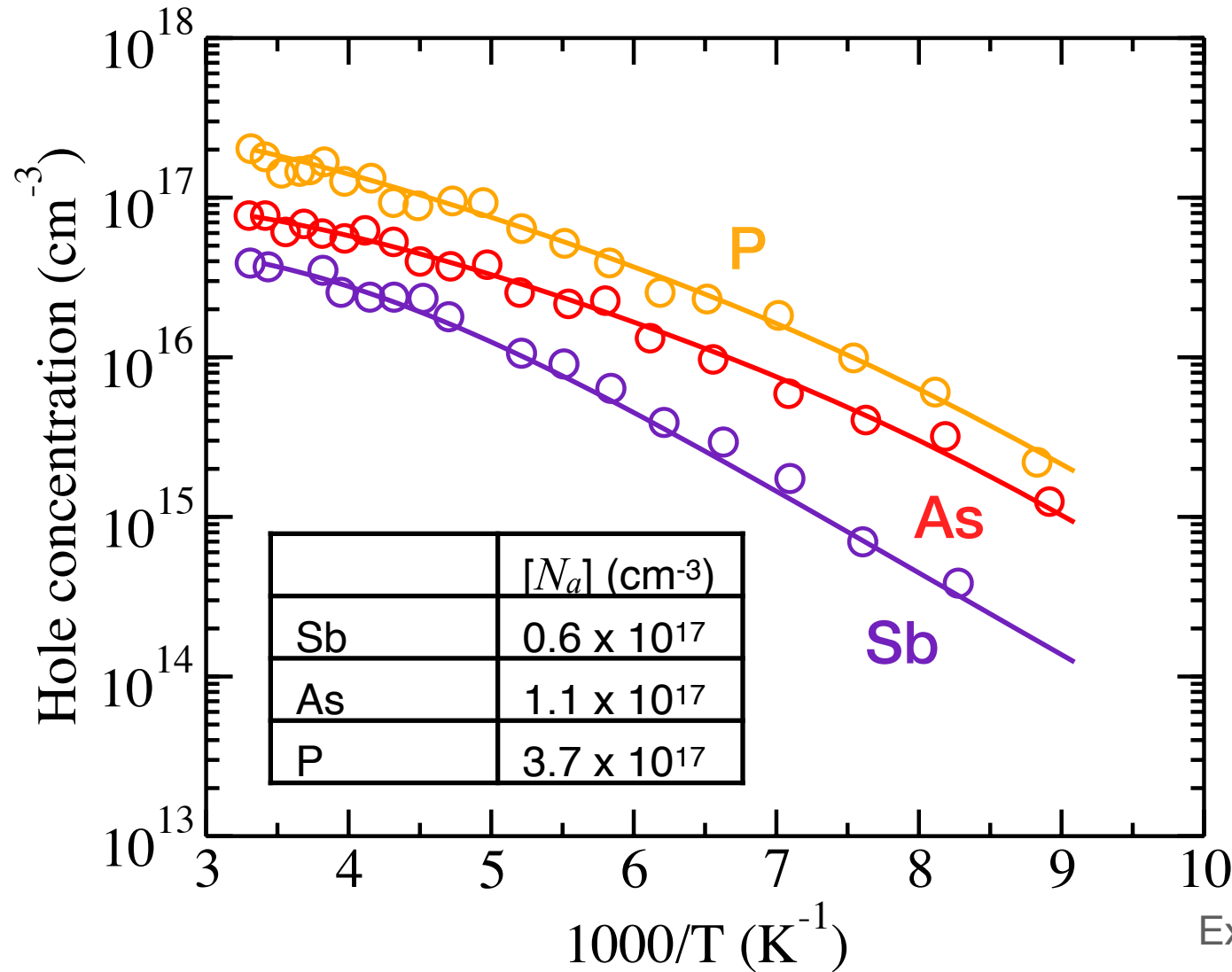
Calculated hole concentration for Sb-doped CdTe



Changing E_a from 116 meV to 150 meV leads to overall lower hole concentration
→ cannot explain experimental data over the whole temperature range

Exp. data:
Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for P, As, and Sb-doped CdTe



Using calc. E_a

P(0/-) = 93 meV

As(0/-) = 99 meV

Sb(0/-) = 116 meV

For P and As, using $[N_d] = 6\% [N_a]$
gives best fit

Source of the compensating donor in
the case of P and As still unknown

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Summary

- Need to include spin-orbit coupling and use very large supercell to describe shallow acceptor centers in CdTe
- P, Sb, As in CdTe are shallow acceptors with ionization energies ~ 100 meV, around that of the hydrogen model
- AX center do not play a role as self-compensation center, expect perhaps in the case of Sb under high doping levels
- Best fit to the exp. data of P and As-doped single crystals indicate presence of compensating donors with 6% of the dopant concentration
- Doping efficiency in single crystals decreases at higher doping ($>10^{17}$ cm $^{-3}$), the cause of which is still unknown
- Low doping efficiency in thin films likely to have contribution from grain boundaries that serve as source or sink of compensating defects.

Chatratin *et al.*, J. Phys. Chem. Lett. **14**, 273 (2023)