

Effects of spin-orbit coupling and very large supercells on the description of acceptor states in CdTe

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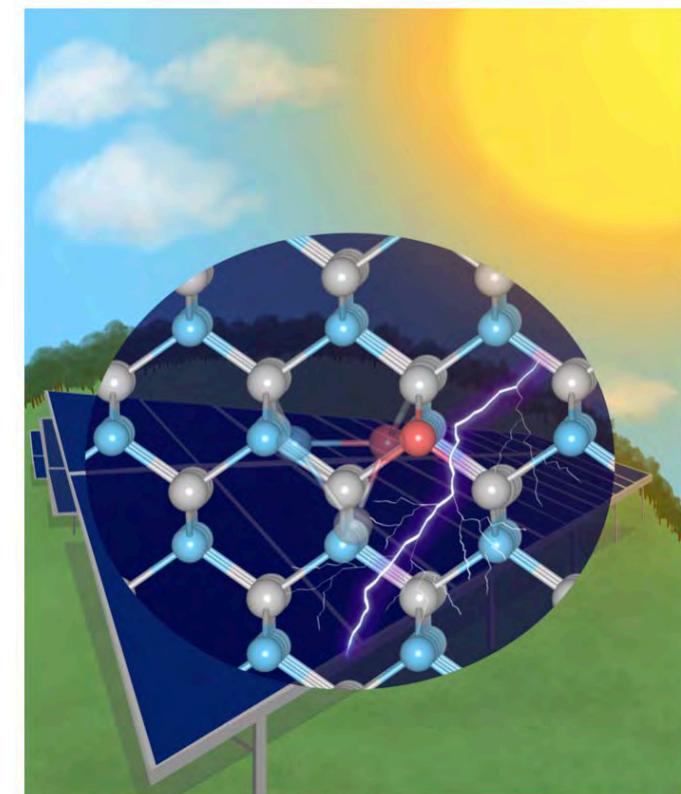
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Award # DE-EE0009344



The 32nd ICDS

International Conference on Defects
in Semiconductors

**September
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Topics: Theory and experiments on
Wide-band-gap oxides and nitrides
Quantum defects, qubits, single-photon emitters
Conventional IV-IV, III-V, and II-VI semiconductors
Complex oxide and halide perovskites
Electronics, optoelectronics, photovoltaic, thermoelectric, magnetic,
ferroelectric, and spintronic materials, radiation detectors
2D materials
...

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**Abstract submission
will be open soon**

Atlantic Sands Hotel
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For more information:



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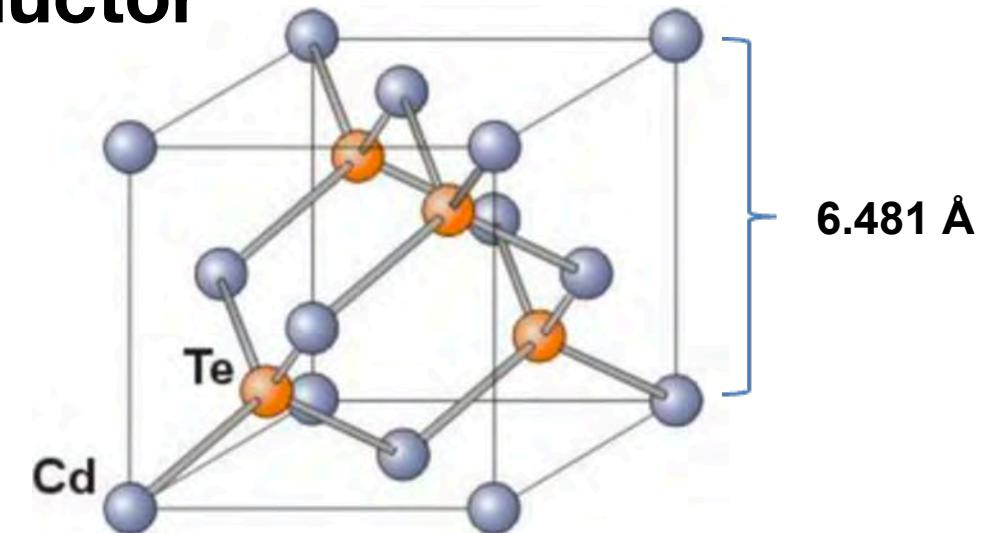


CdTe semiconductor

member of the II-VI family

zinc blende crystal structure

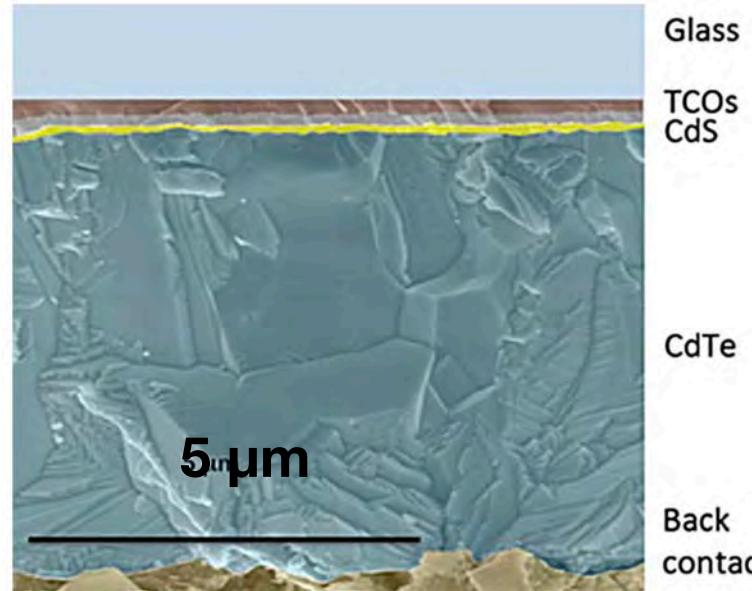
direct band gap = 1.5 eV



single crystals



thin films vapor transport deposition



Applications:

thin film solar cells

infrared optical windows/lenses

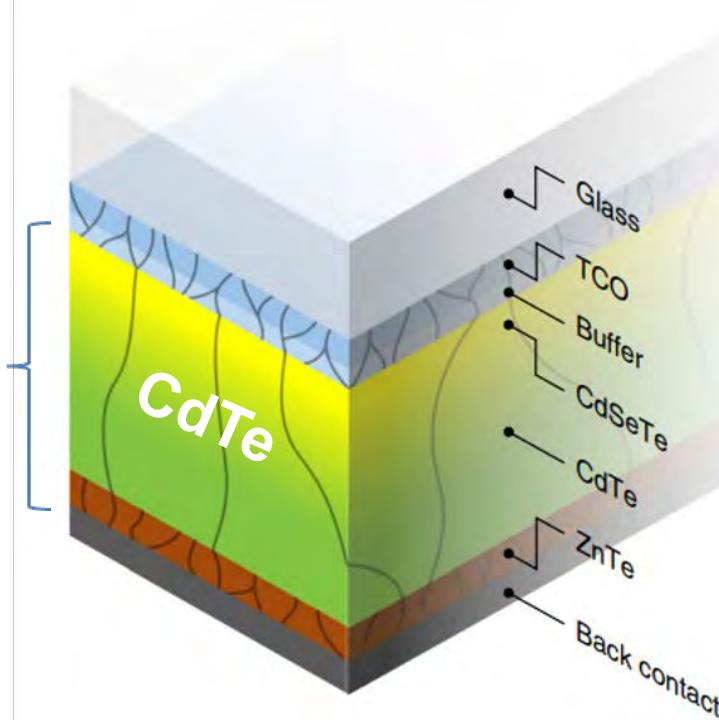
electro-optic modulators

scintillators

Matt Reese (NREL)

CdTe in solar cells

thin-film photovoltaic technology



Polycrystalline

→ grain boundaries

vapor transport deposition
→ low production cost

Metzger et al. Nat. Energy 4, 837 (2019)

record efficiency in the lab cells: 22.3%

Commercially available

19.3%
HIGH BIN EFFICIENCY

30YR
LINEAR PERFORMANCE WARRANTY

Series 7

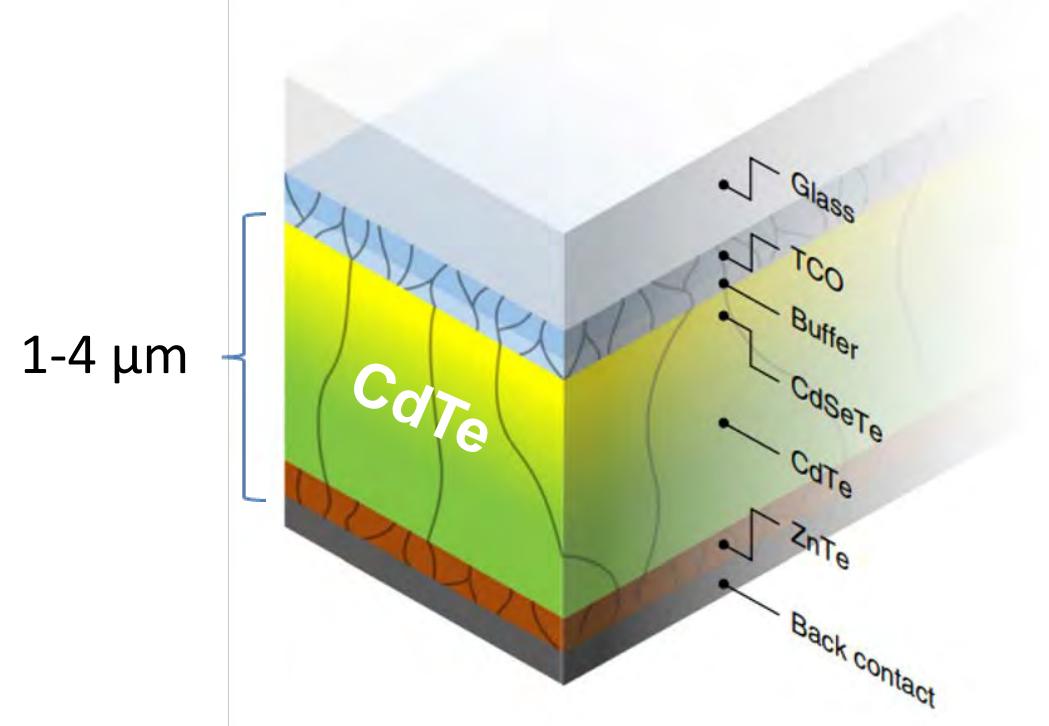


Made using same process to fabricate modules

CdTe in solar cells

thin-film photovoltaic technology

Typical minority-carrier device



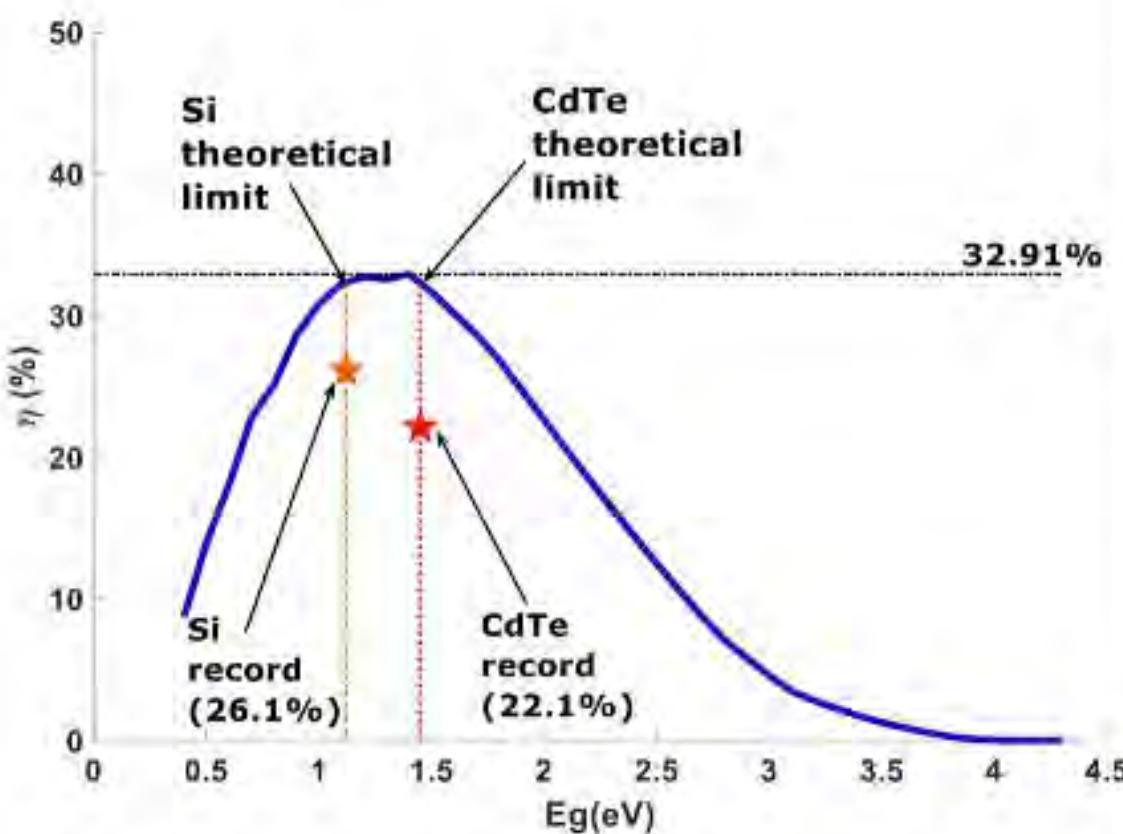
Metzger et al. Nat. Energy 4, 837 (2019)

Current limitations:

- low open-circuit voltage V_{oc} ($\sim 0.8\text{-}0.9$ eV) $\ll E_g$ (1.5 eV)
- undoped, low hole concentrations $\sim 10^{14} \text{ cm}^{-3}$
- layers doped with As show very low doping efficiency
 $\rightarrow [\text{As}] \sim 10^{18} \text{ cm}^{-3}$, $[\text{holes}] \sim 10^{16} \text{ cm}^{-3}$
where do the dopants go?
- short minority carrier lifetimes
- polycrystalline films
 \rightarrow carrier recombination in the bulk and at grain boundaries

CdTe solar cells - room for improvement

Current efficiencies are well below the theoretical limit of 33%



Device modeling indicate that efficiency of 25% can be achieved if:

hole concentration $> 10^{16} \text{ cm}^{-3}$
while keeping everything else the same

carrier lifetime $\geq 100 \text{ ns}$,
interface recombination velocity $\leq 1000 \text{ cm/s}$

Burst *et al.*, Nature Energy 1, 16015 (2016).

Kanevce, *et al.*, J. Appl. Phys. 121, 214506 (2017).

Back to the basics of doping CdTe

Periodic table of the elements

period	group	Alkali metals		Halogens		Alkaline-earth metals		Noble gases		Transition metals		Rare-earth elements (21, 39, 57–71) and lanthanoid elements (57–71 only)		Other metals		Other nonmetals		Actinoid elements		He	
1	1*	H																			
1	2		Li	Be																He	
2	3	Na	Mg																	Ne	
3	11	Al	Si																	Ar	
4	19	K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	S	Br	Kr		
5	37	Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Au	Cd	In	Sn	St	Te	I	Xe		
6	55	Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn		
7	87	Fr	Ra	Ac	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Nh	Fl	Mc	Lv	Ts	Og		
lanthanoid series		58	59	60	61	62	63	64	65	66	67	68	69	70	71						
actinoid series		Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu						
		90	91	92	93	94	95	96	97	98	99	100	101	102	103						
		Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr						

For p-type doping:

Look to the left of Cd or Te

*Numbering system adopted by the International Union of Pure and Applied Chemistry (IUPAC). © Encyclopædia Britannica, Inc.

Back to the basic aspects of doping CdTe

Periodic table of the elements

Periodic table of the elements showing the IUPAC numbering system. The table includes groups 1 and 2, transition metals, rare-earth elements, and actinoid elements. Specific elements highlighted include Cu, Ag, Cd, Te, Sb, Bi, and Hg. The lanthanoid and actinoid series are also shown.

group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18	
1	H												B	C	N	O	F	He	
2	Li	Be											Al	Si	P	S	Cl	Ne	
3	Na	Mg	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	30	Zn	31	32	33	34	35	36
4	K	Ca	21	22	23	24	25	26	27	28	47	48	33	As	Ge	Se	Br	Kr	
5	Rb	Sr	39	40	41	42	43	44	45	46	Ag	Cd	49	50	51	52	53	54	
6	Cs	Ba	La	Hf	Ta	W	75	76	77	78	79	80	In	Sn	Sb	Te	I	Xe	
7	Fr	Ra	89	104	105	106	107	108	109	110	111	112	113	114	115	116	117	118	
				Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Nh	Fl	Mc	Lv	Ts	Og		
lanthanoid series	6	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu				
actinoid series	7	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr				

*Numbering system adopted by the International Union of Pure and Applied Chemistry (IUPAC). © Encyclopædia Britannica, Inc.

For p-type doping:

Look to the left of Cd or Te

Try and minimize chemical and size mismatches

Cu, Ag on the Cd site

P, As, or Sb on the Te site

Typical experimental data on doping p-type of CdTe

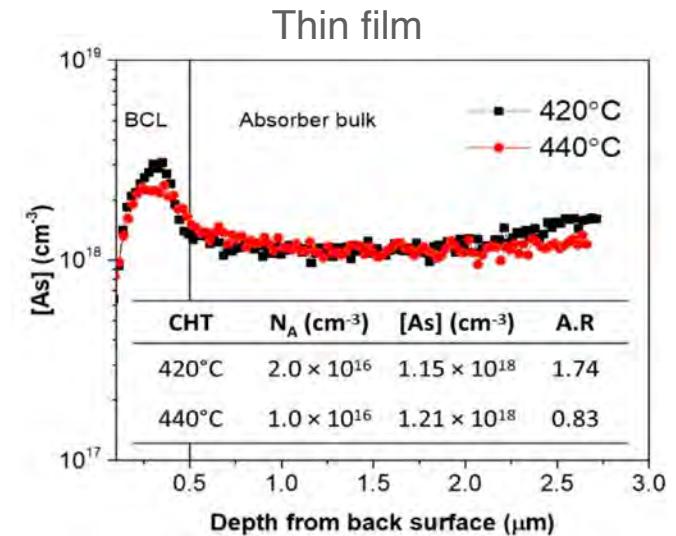
Cu-doped CdTe absorber layers $< 10^{15} \text{ cm}^{-3}$

+ stability issues → Cu interstitials are highly mobile

Corwine *et al.*, *Sol. Energy Mater. Sol. Cells* 82, 481 (2004)

Grecu, *et al.*, *J. Appl. Phys.* 88, 2490 (2000)

Burst *et al.*, *APL Mater.* 4, 116102 (2016)



As, P, Sb doping: $10^{15} - 10^{16}$ holes/cm⁻³

very low doping efficiency, [free carrier] << [dopant]

highly compensated

+ short carrier lifetimes

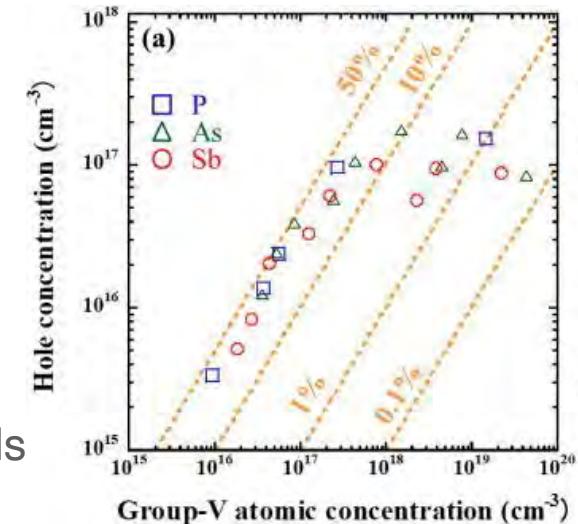
McCandless *et al.*, "IEEE J. Photovolt." 9, 912 (2019)

Metzger *et al.*, *Nature Energy*, 4 837 (2019)

Kartopuetal., *Sol. Energy Mater. Sol. Cells* 194, 259 (2019)

Source of compensation unknown!
Dopants in the grain boundaries?

Oklobia *et. al.* IEEE J. Photovolt. 12, 1296 (2022)



Single crystals

Nagaoka *et. al.*, *Appl. Phys. Lett.* 116, 132102 (2020)

Sb, As and P doping in CdTe single crystals

Temperature dependent Hall data

Partially compensated acceptors

$$p = -A + \sqrt{A^2 + \frac{N_V}{2} (N_A - N_D) \exp\left(-\frac{E_a}{k_B T}\right)}$$

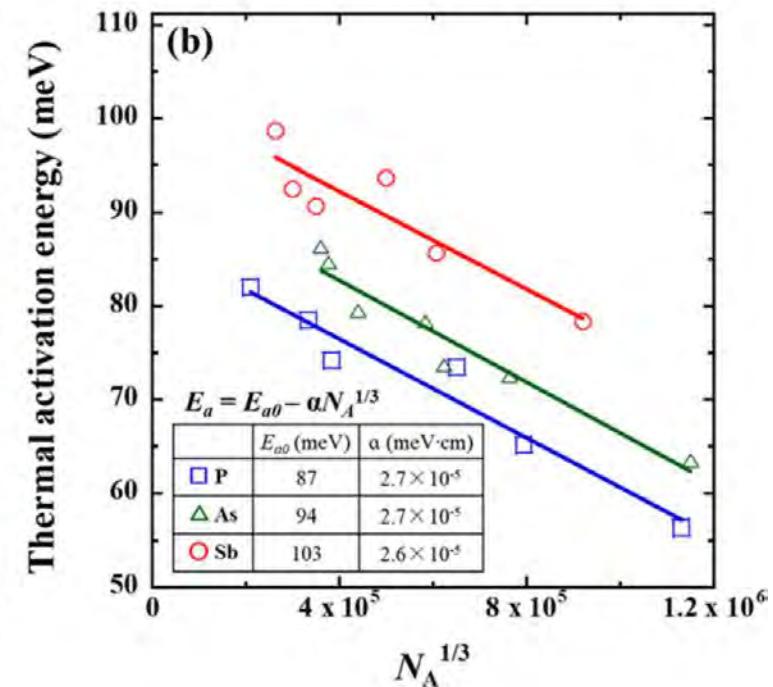
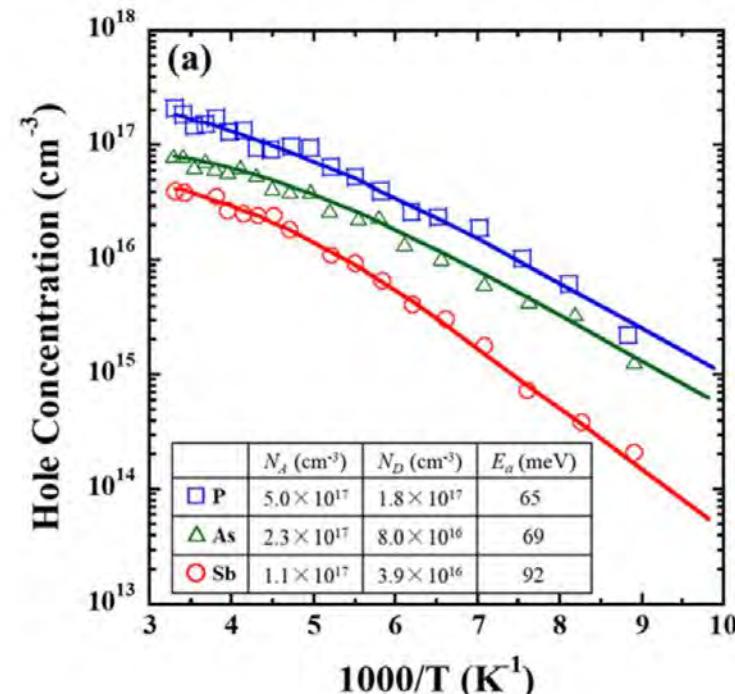
$$A = \frac{1}{2} \left[N_D + \frac{N_V}{2} \exp\left(-\frac{E_a}{k_B T}\right) \right]$$

Blakemore, Semiconductor statistics
(Courier Corp., 2002)

Data fit to obtain E_a , N_A , ...

	N_A (cm ⁻³)	N_D (cm ⁻³)	E_a (meV)
□ P	5.0×10^{17}	1.8×10^{17}	65
△ As	2.3×10^{17}	8.0×10^{16}	69
○ Sb	1.1×10^{17}	3.9×10^{16}	92

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)



$$E_a = E_{a0} - a N_A^{1/3}$$

	E_{a0} (meV)	a (meV·cm)
□ P	87	2.7×10^{-5}
△ As	94	2.7×10^{-5}
○ Sb	103	2.6×10^{-5}

dilute limit

What do we know about acceptor impurities in CdTe from theory

Te_i $\frac{0.57}{}$ (0/-2)

V_{Cd} $\frac{0.21}{\frac{0.13}{}}$ (-2-) $\frac{0.22}{\frac{0.20}{\frac{0.15}{}}}$ (0/-) $\frac{Cu_{Cd}}{Au_{Cd}}$ $\frac{Ag_{Cd}}{Na_{Cd}}$ $\frac{0.02}{}$

DFT-LDA
LAPW, non-relativistic
Supercells with 32 atoms

Bi_{Te} $\frac{0.30}{}$
 Sb_{Te} $\frac{0.23}{}$
 As_{Te} $\frac{0.10}{}$ (0/-)
 P_{Te} $\frac{0.05}{}$
 N_{Te} $\frac{0.01}{}$

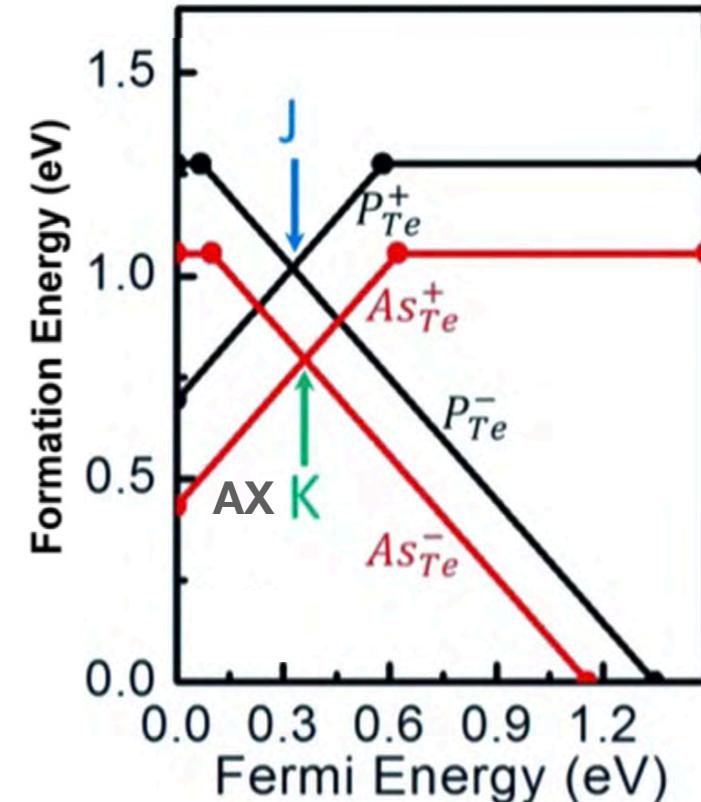
VBM

Wei and Zhang, Phys. Rev. B **66**, 155211 (2002)

These predictions indicate that it
Would be impossible to make
CdTe p-type with As or P

P and As are shallow acceptors
Self-compensation by AX centers
Fermi level pinned in the gap
negligible hole concentration

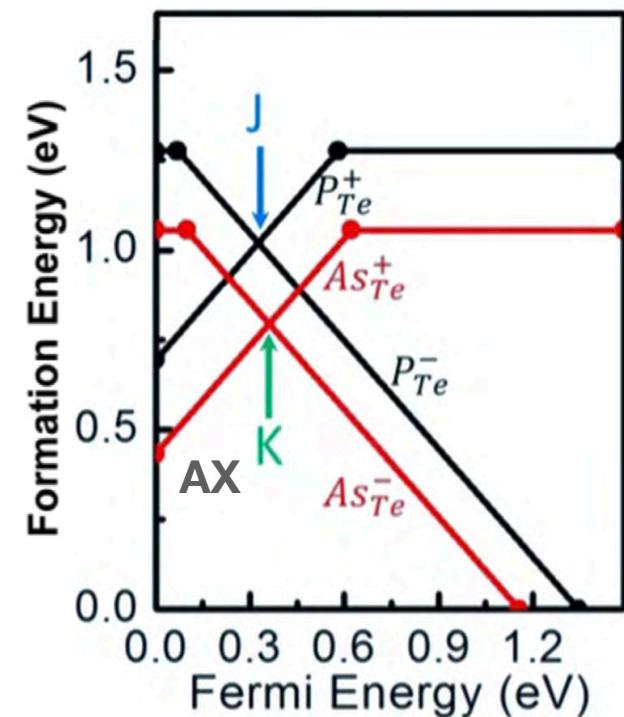
HSE06 hybrid functional
Supercell 64 atoms
no spin-orbit



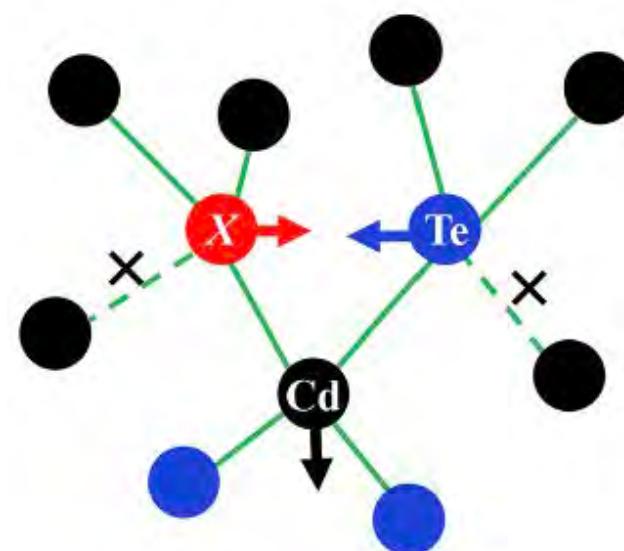
Yang et. al., Semicond. Sci. Technol. **31**, 083002 (2016)

What do we know about acceptor impurities in CdTe from theory

HSE06 hybrid functional
Supercell 64 atoms
no spin-orbit

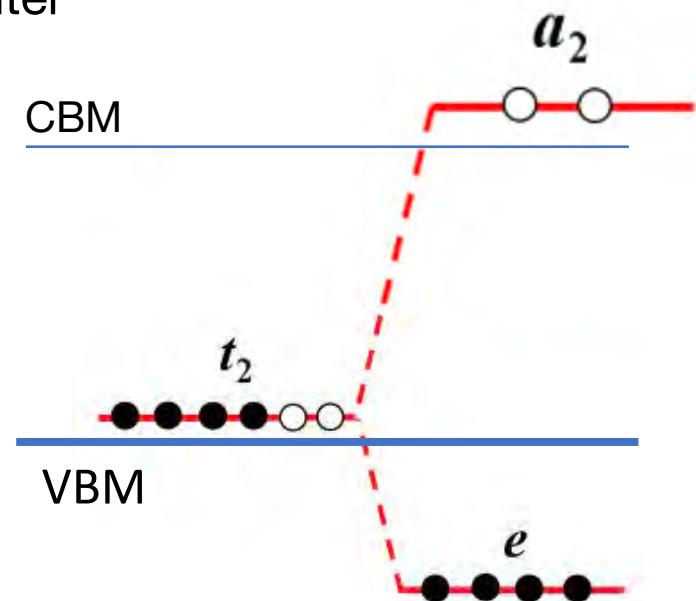


P and As are shallow acceptors
Self-compensation by AX centers
Fermi level pinned in the gap
negligible hole concentration



Chadi, Phys. Rev. B 59, 15181 (1999)

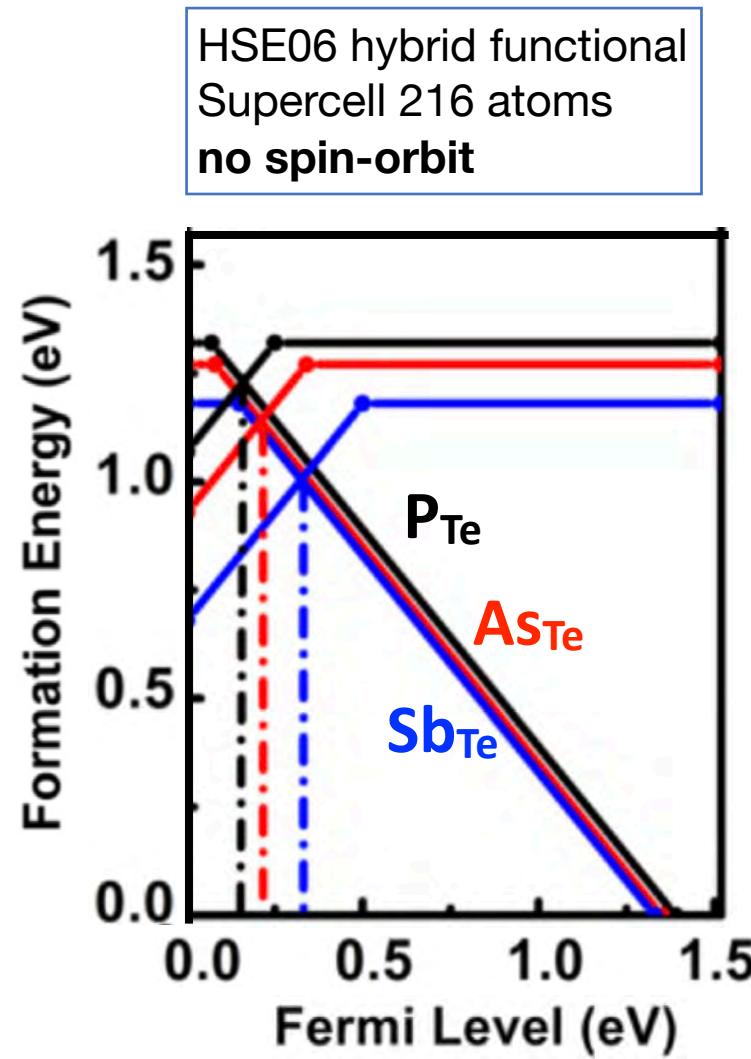
AX center



Yang et. al., Semicond. Sci. Technol. 31, 083002 (2016)

Large local lattice relaxation
Breaking two bonds and forming a X_{Te} -Te bond

What do we know about acceptor impurities in CdTe from theory



(0/-) transition levels

P	70 meV
As	80 meV
Sb	150 meV

Fermi level pinned at the (+/-) level
→ negligible hole concentration
→ full self-compensation by AX centers

Had to use arguments based nonequilibrium
or kinetics to explain observed hole concentrations

Te is a heavy atom → large splitting at the top of the valence band expected due to spin-orbit coupling

What are the effect of SOC?

band structure → push up the valence-band maximum (VBM)

defect levels

defect formation energies

Acceptor ionization energies and dependence on the supercell size

Stability of the AX centers

CdTe basic properties, different functionals

	PBEsol		HSE06		HSE(0.33)		Exp.
	no SOC	with SOC	no SOC	with SOC	no SOC	with SOC	
a (Å)	6.471		6.569		6.545		6.481
E_g (eV)	0.667	0.397	1.500	1.197	1.814	1.502	1.513 (T=300K)
ΔH_f (eV)	0.759		1.147		1.307		-0.96 -1.17

Sirdeshmukh *et al.*, Cryst. Res. Technol. **28**, 15 (1993)

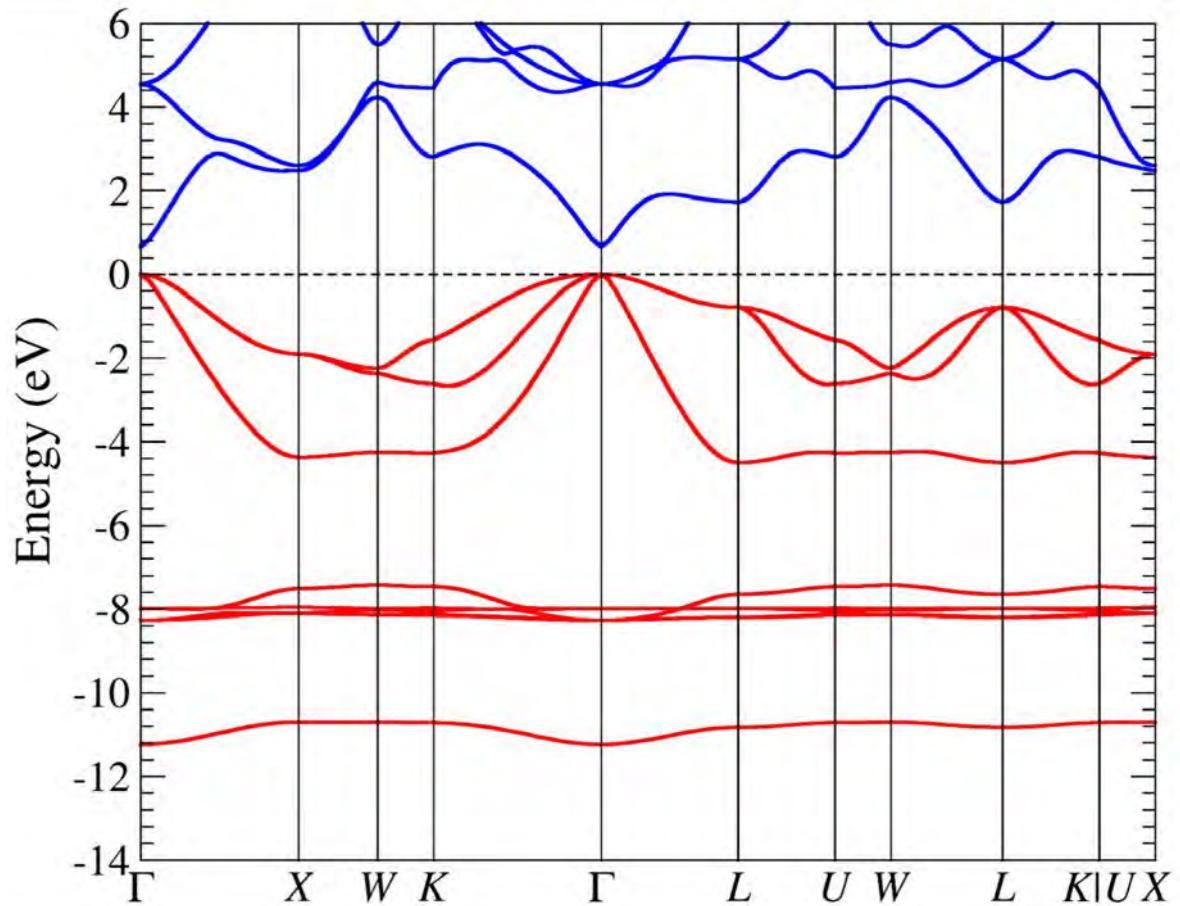
Fonthal *et al.*, J. Phys. Chem. Solids **61**, 579 (2000)

Dean, Lange's Handbook of Chemistry (McGraw–Hill, New York, 1999)

Yamaguchi *et al.*, Materials transactions, JIM **41**, 790 (2000)

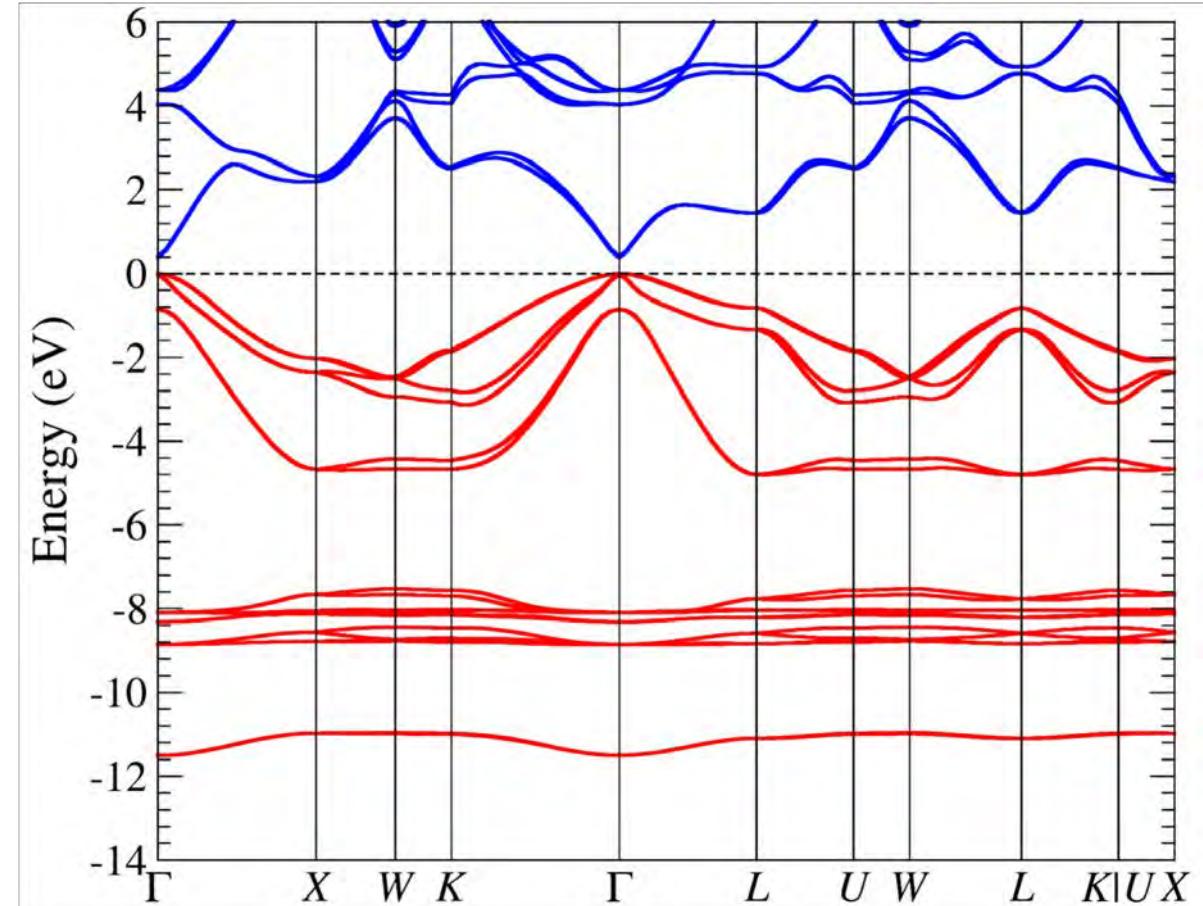
CdTe electronic structure, with and without SOC

DFT-GGA (PBEsol)



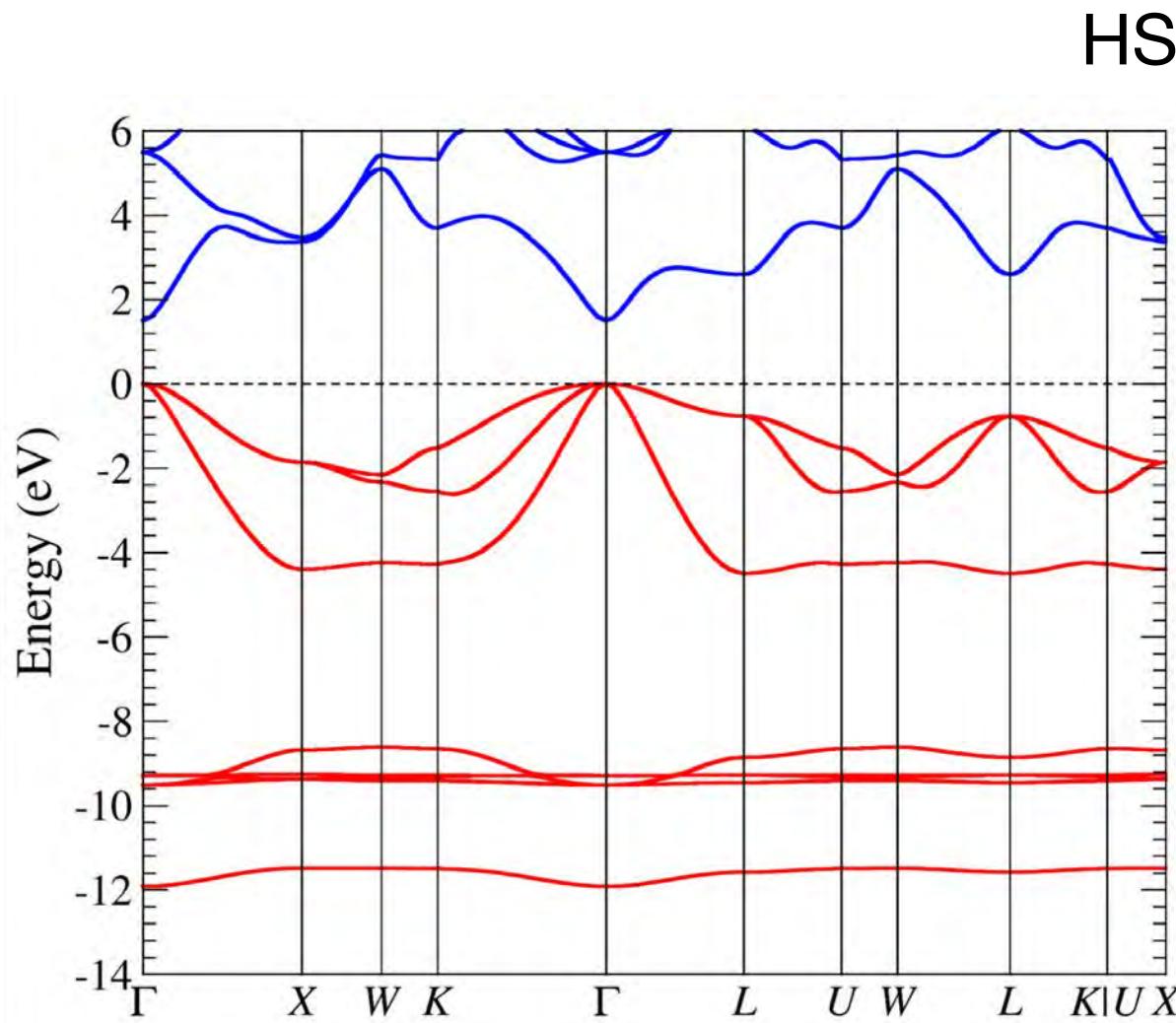
Band gap = 0.667 eV

with SOC

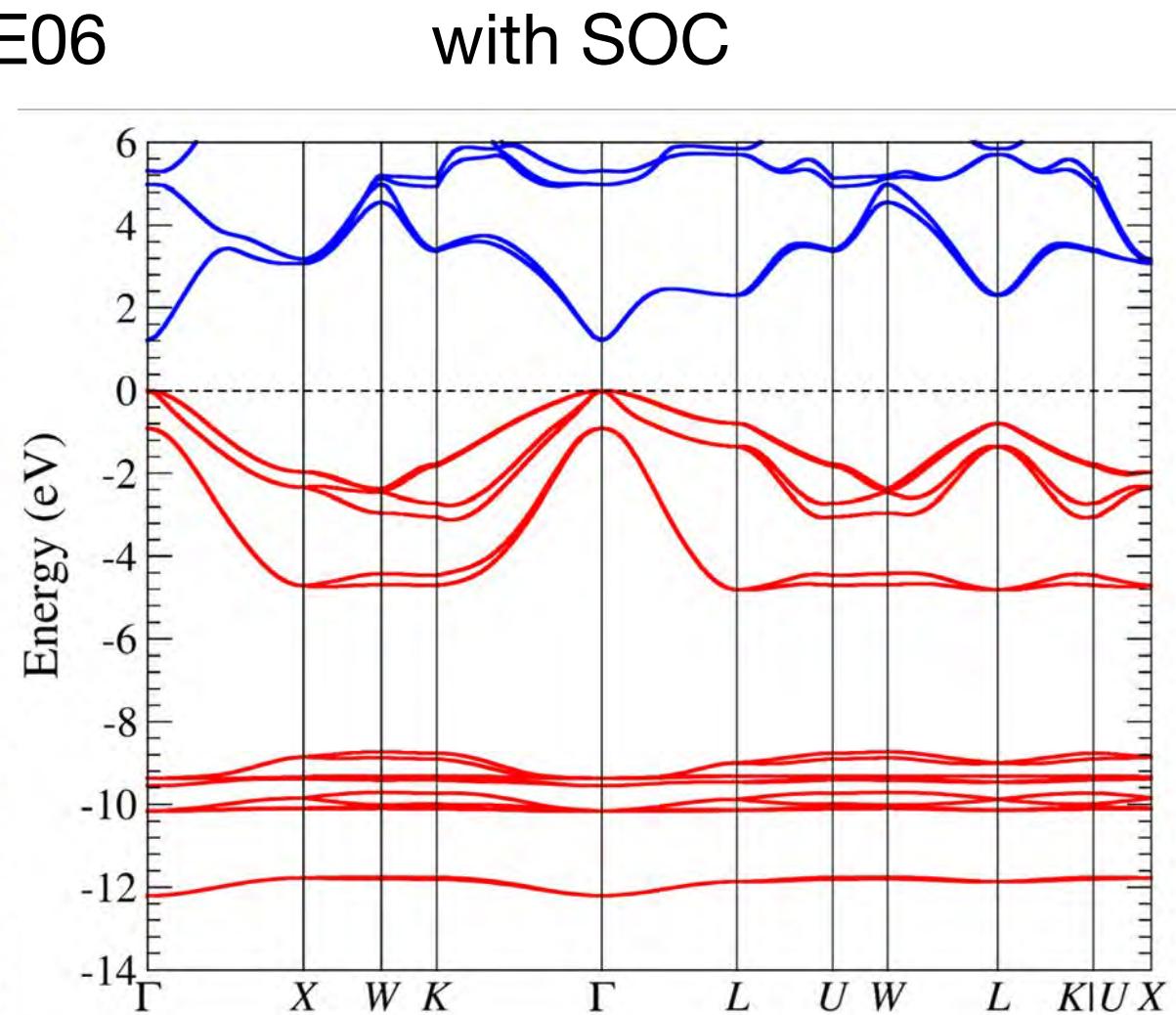


Band gap = 0.397 eV

CdTe electronic structure, with and without SOC

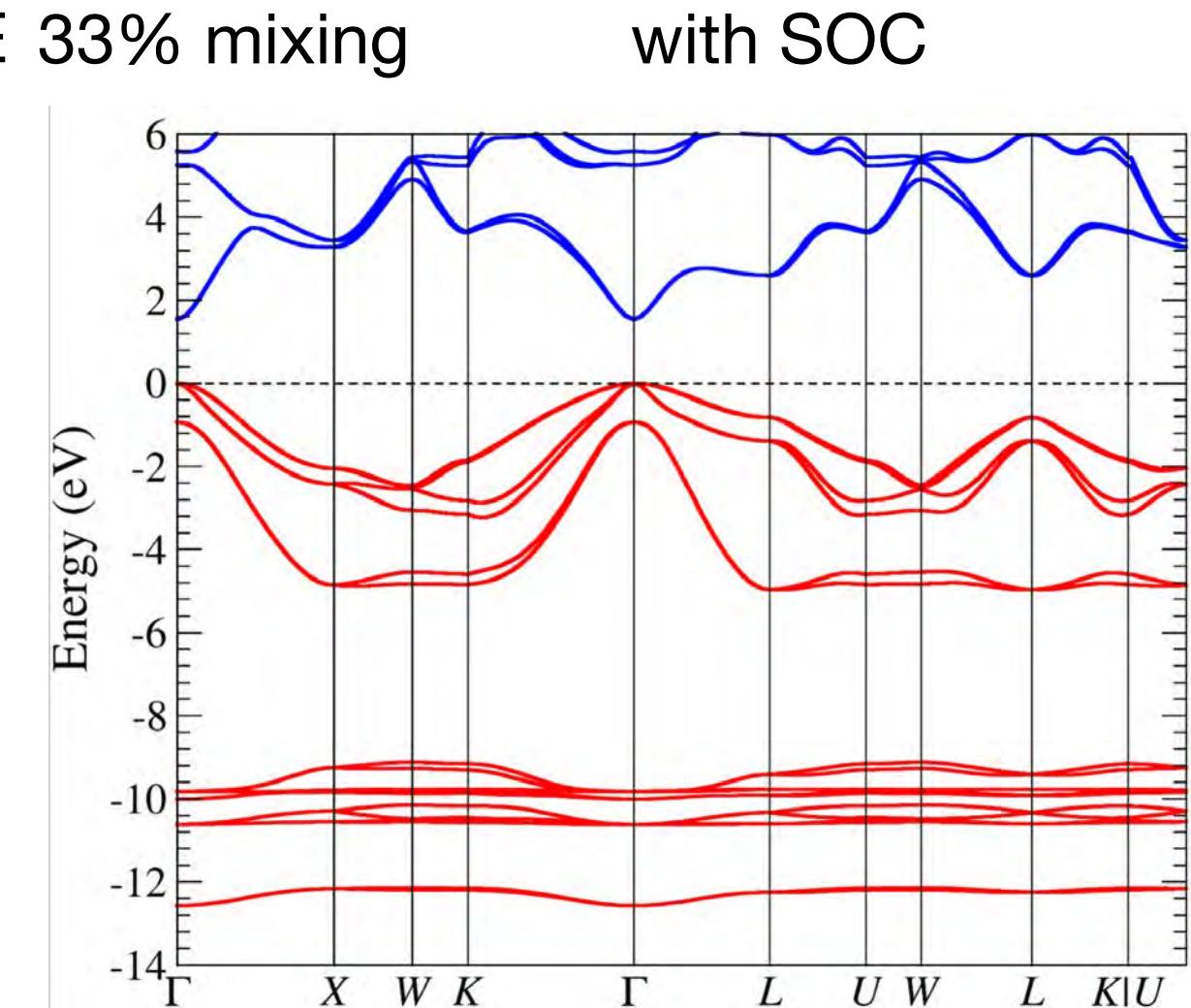
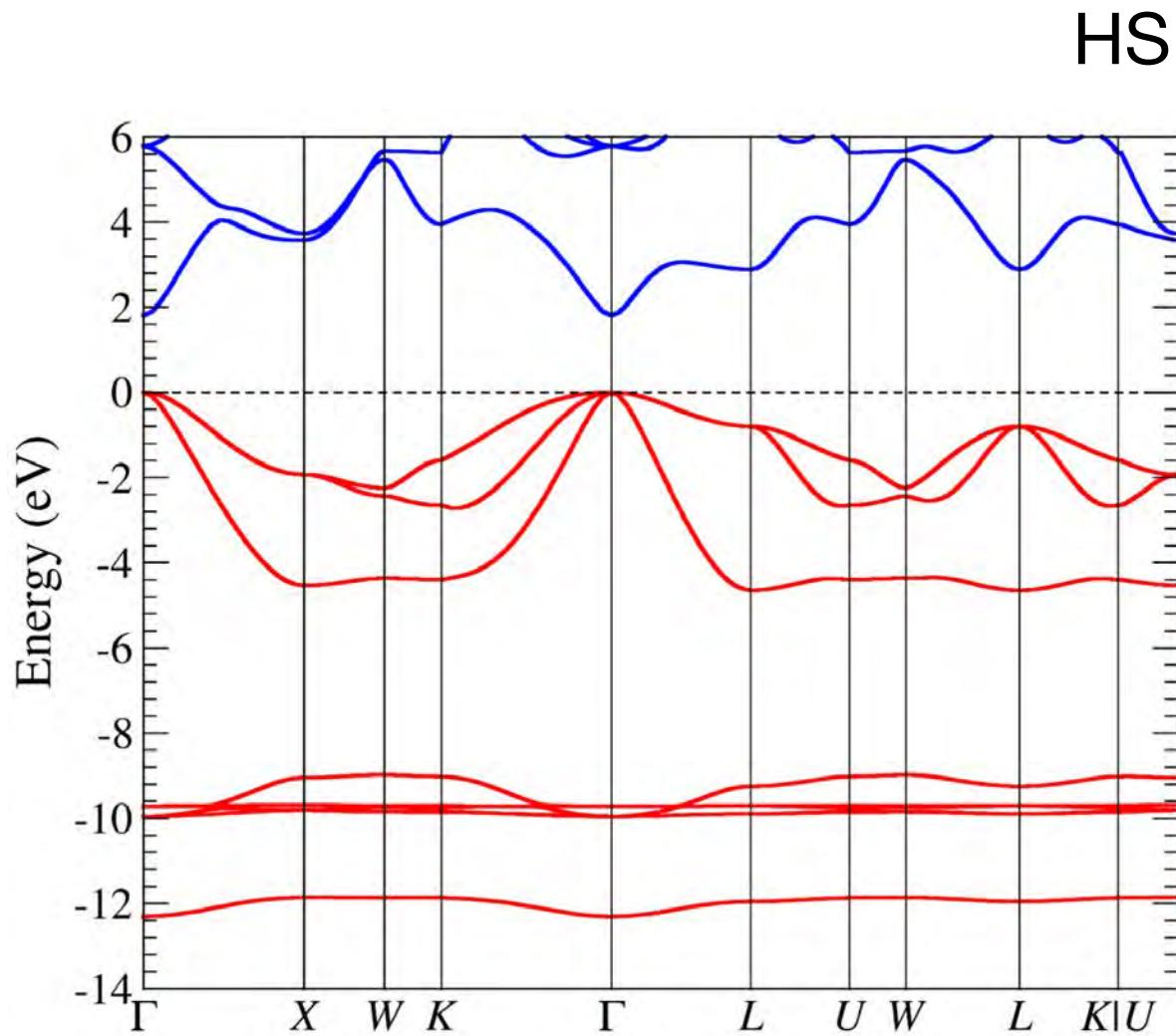


Band gap = 1.500 eV



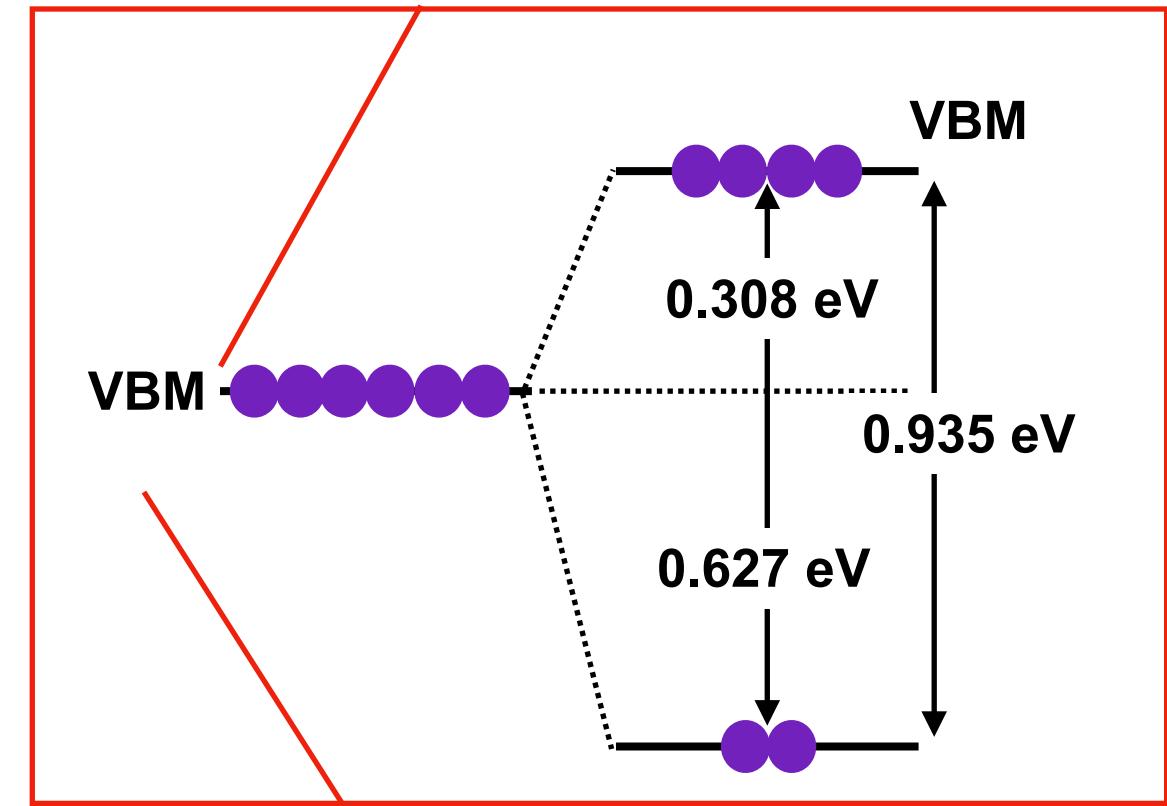
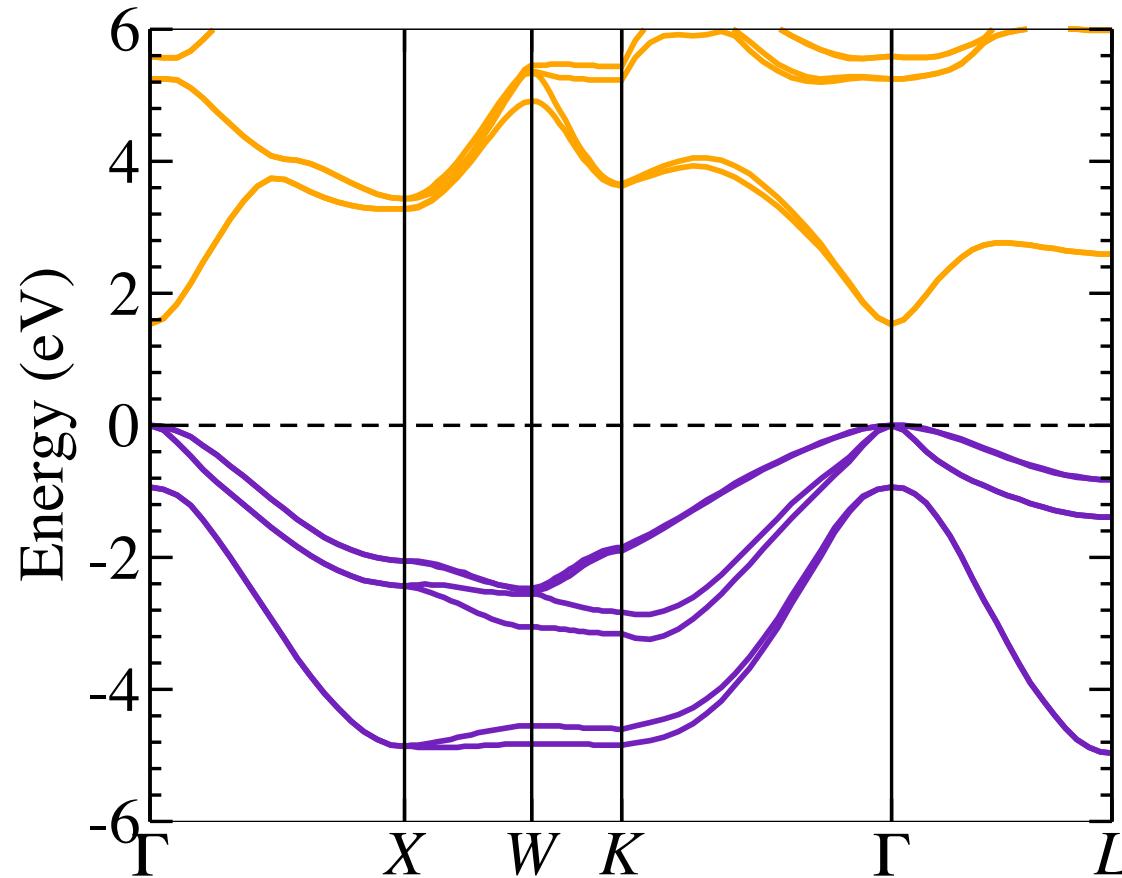
Band gap = 1.197 eV

CdTe electronic structure, with and without SOC

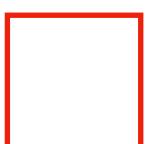


CdTe electronic structure HSE $\alpha = 0.33$

spin-orbit splitting



see also Pan *et al.*, Phys. Rev. B **98** 054108 (2018)



CdTe effective masses

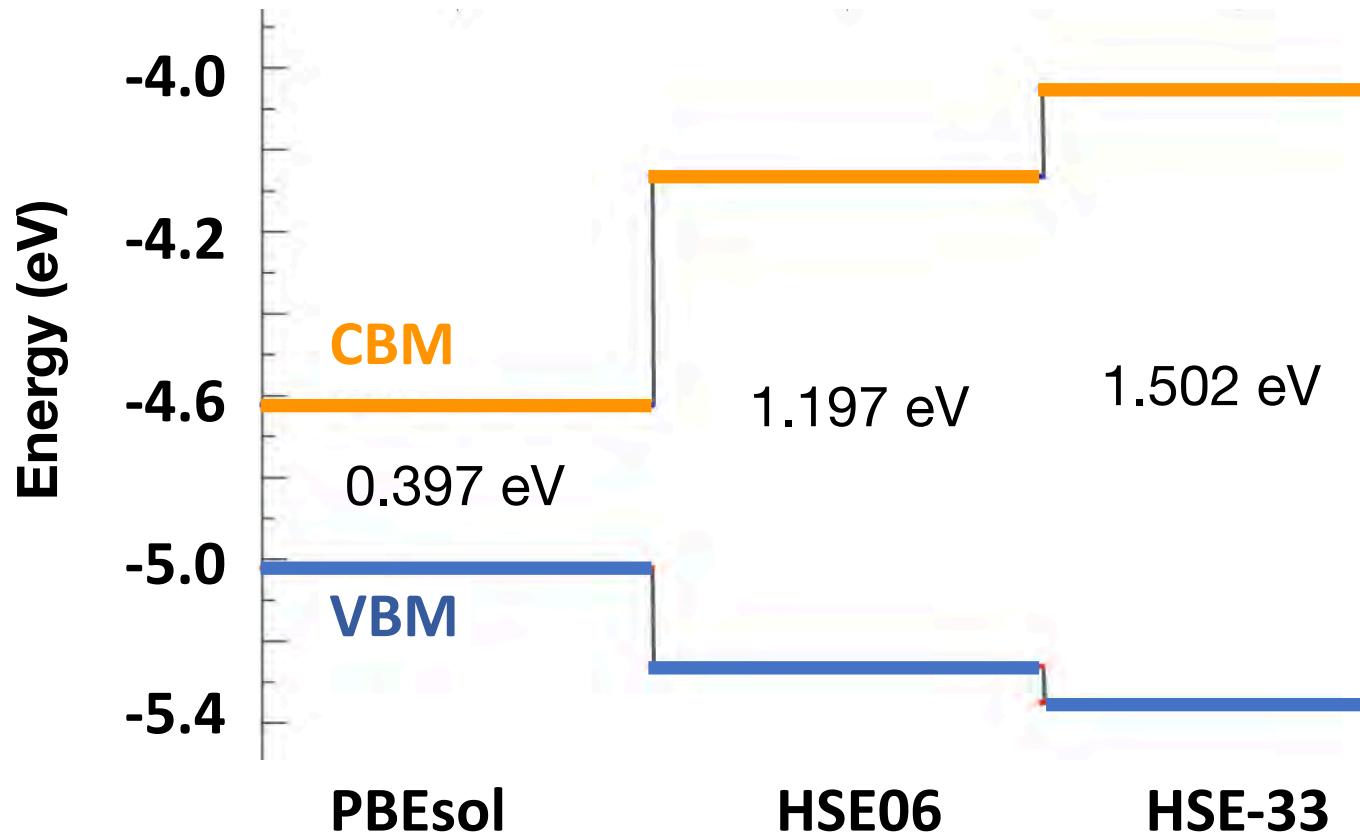
HSE $\alpha = 0.33$

	PBEsol		HSE06		HSE(0.33)		Exp.
Effective mass	no SOC	with SOC	no SOC	with SOC	no SOC	with SOC	
[100] G-X	$m_{hh}= 0.504$ $m_{lh}= 0.064$ $m_e= 0.063$	$m_{hh}= 0.440$ $m_{lh}= 0.060$ $m_{sp}= 0.246$ $m_e= 0.053$	$m_{hh}= 0.533$ $m_{lh}= 0.103$ $m_e= 0.100$	$m_{hh}= 0.474$ $m_{lh}= 0.113$ $m_{sp}= 0.307$ $m_e= 0.094$	$m_{hh}= 0.522$ $m_{lh}= 0.113$ $m_{sp}= 0.321$ $m_e= 0.110$	$m_{hh}= 0.471$ $m_{lh}= 0.129$ $m_{sp}= 0.321$ $m_e= 0.106$	$m_{hh}= 0.72$ $m_{lh}= 0.1$ $m_e= 0.094$
[110] G-K	$m_{hh}= 2.586, 0.502$ $m_{lh}= 0.060$ $m_e= 0.065$	$m_{hh}= 0.797$ $m_{lh}= 0.060$ $m_{sp}= 0.243$ $m_e= 0.055$	$m_{hh}= 2.581, 0.530$ $m_{lh}= 0.090$ $m_e= 0.102$	$m_{hh}= 0.865$ $m_{lh}= 0.105$ $m_{sp}= 0.303$ $m_e= 0.096$	$m_{hh}= 2.612, 0.520$ $m_{lh}= 0.097$ $m_e= 0.111$	$m_{hh}= 0.860$ $m_{lh}= 0.117$ $m_{sp}= 0.317$ $m_e= 0.108$	$m_{hh}= 0.81$ $m_{lh}= 0.12$ $m_e= 0.096$
[111] G-L	$m_{hh}= 1.083$ $m_{lh}= 0.056$ $m_e= 0.062$	$m_{hh}= 1.112$ $m_{lh}= 0.055$ $m_{sp}= 0.244$ $m_e= 0.051$	$m_{hh}= 1.132$ $m_{lh}= 0.085$ $m_e= 0.100$	$m_{hh}= 1.024$ $m_{lh}= 0.101$ $m_{sp}= 0.304$ $m_e= 0.094$	$m_{hh}= 1.127$ $m_{lh}= 0.092$ $m_e= 0.110$	$m_{hh}= 1.191$ $m_{lh}= 0.111$ $m_{sp}= 0.320$ $m_e= 0.106$	$m_e= 0.095$
ionization potential (eV)	-5.022		-5.261		-5.350		-5.6, -5.8

HSE-33% improves the description of effective masses and ionization potential

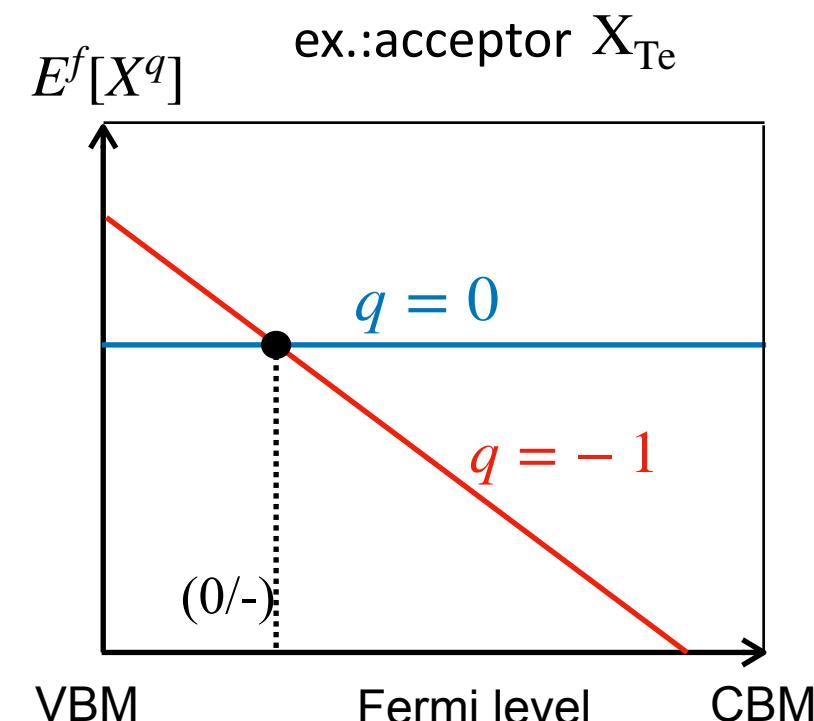
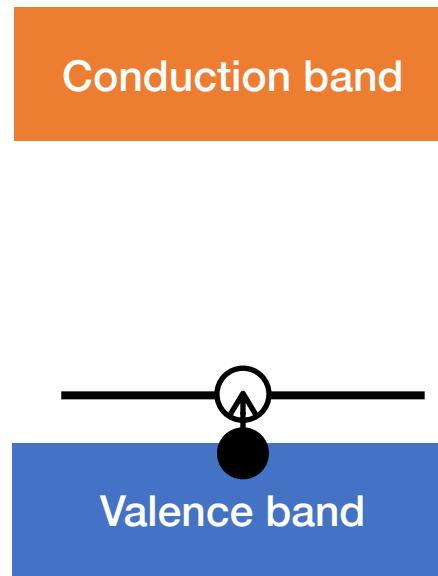
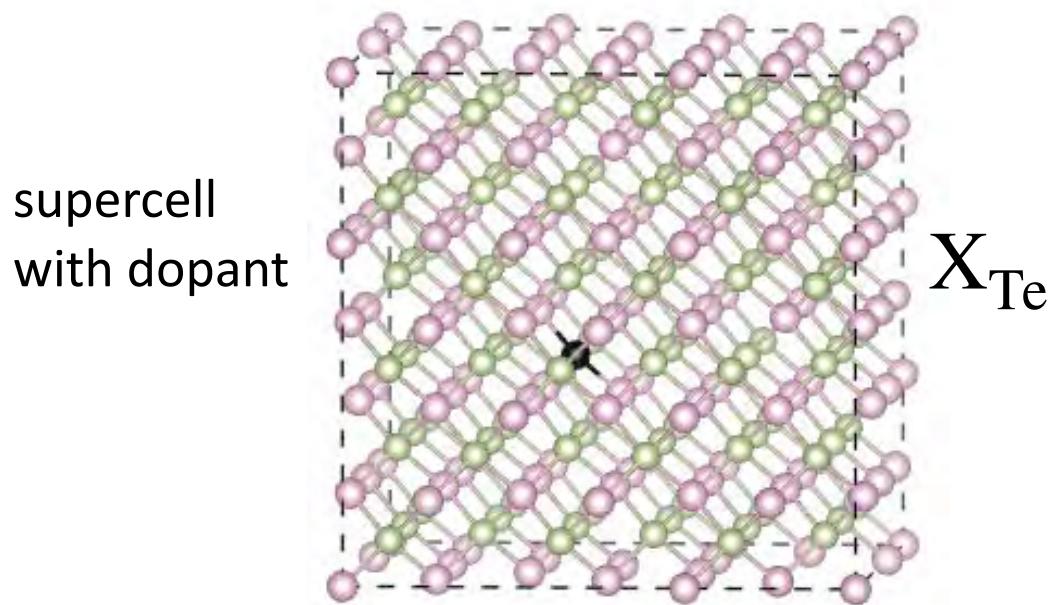
CdTe ionization potential

different functionals



affects both valence and conduction band, almost equally

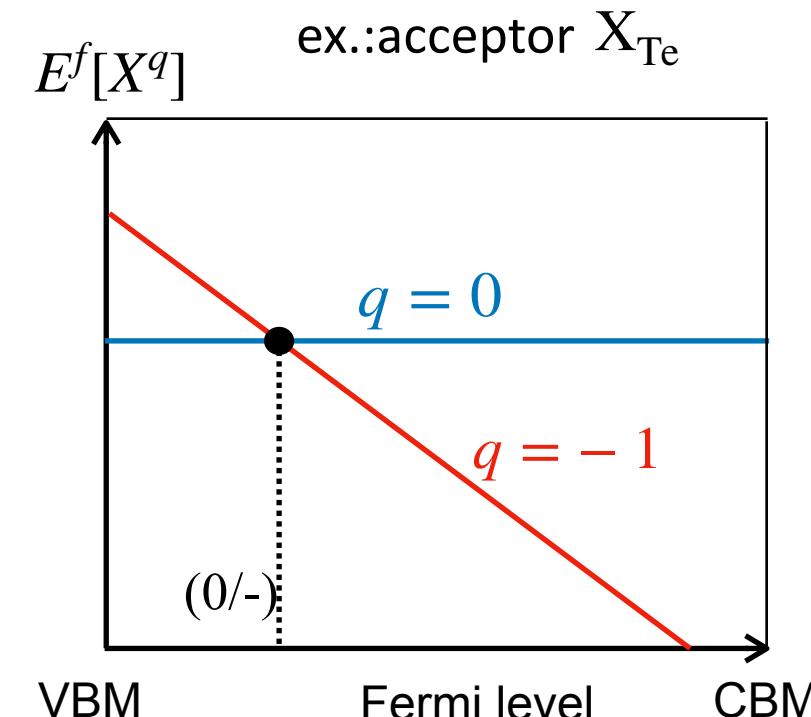
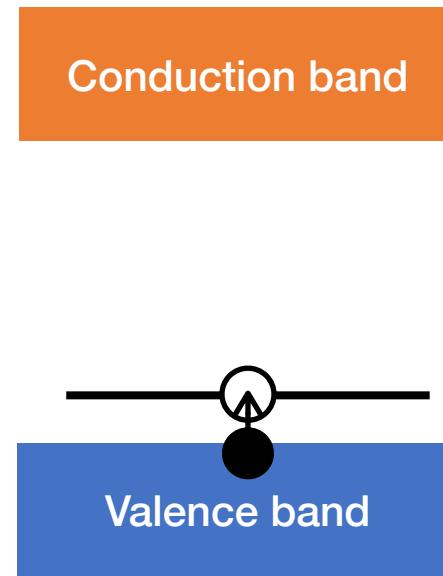
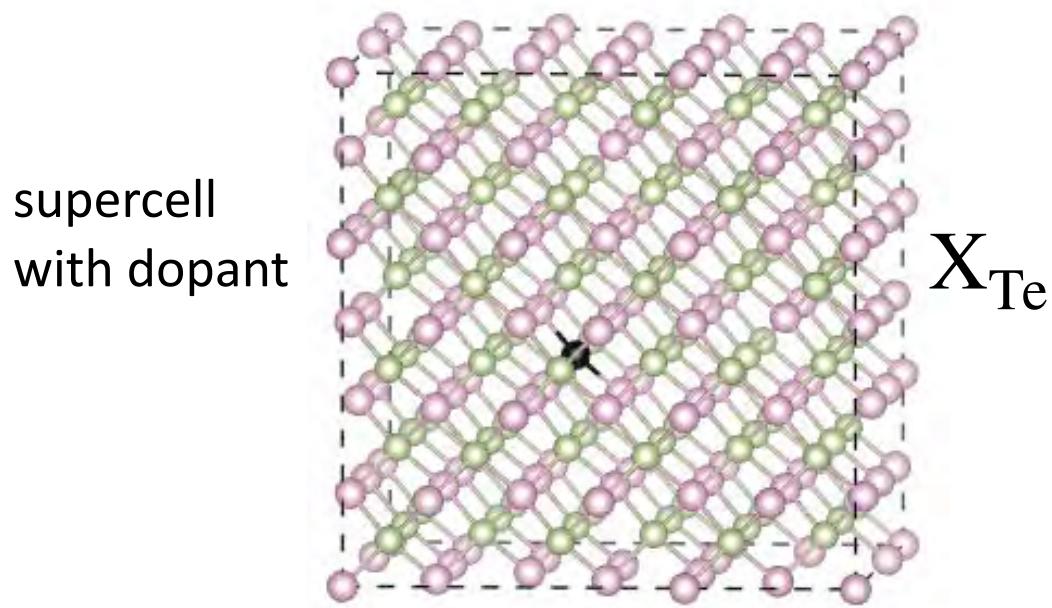
Dopant/defect formation energy and transition level



$$E^f[X^q] = E_{\text{tot}}[X^q] - E_{\text{tot}}[\text{bulk}] + \sum_i n_i \mu_i + q(\varepsilon_f + E_{VBM}) + \Delta^q$$

Freysoldt *et. at.*, Rev. Mod. Phys. 86, 253 (2014)

Dopant/defect formation energy and transition level



$$E^f[X^q] = E_{\text{tot}}[X^q] - E_{\text{tot}}[\text{bulk}] + \sum_i n_i \mu_i + q(\varepsilon_f + E_{\text{VBM}}) + \Delta^q$$

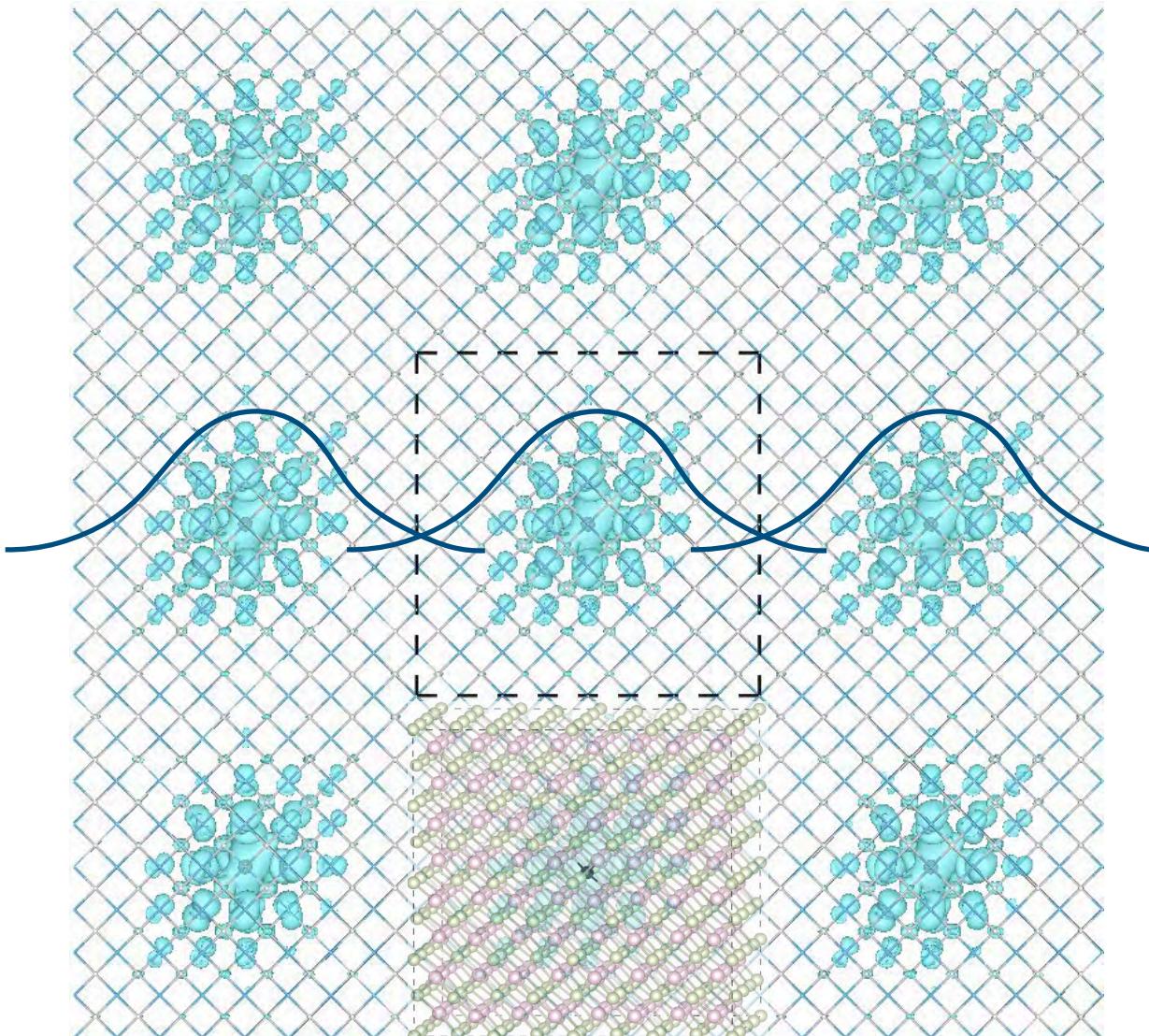
Typical supercell sizes are not large enough to describe an isolated shallow center

Swift *et al.*, Npj Comput. Mater. **6**, 181 (2020)

King and Wang, Phys. Rev. Appl. 18, 064001 (2022)

Freysoldt *et. at.*, Rev. Mod. Phys. **86**, 253 (2014)

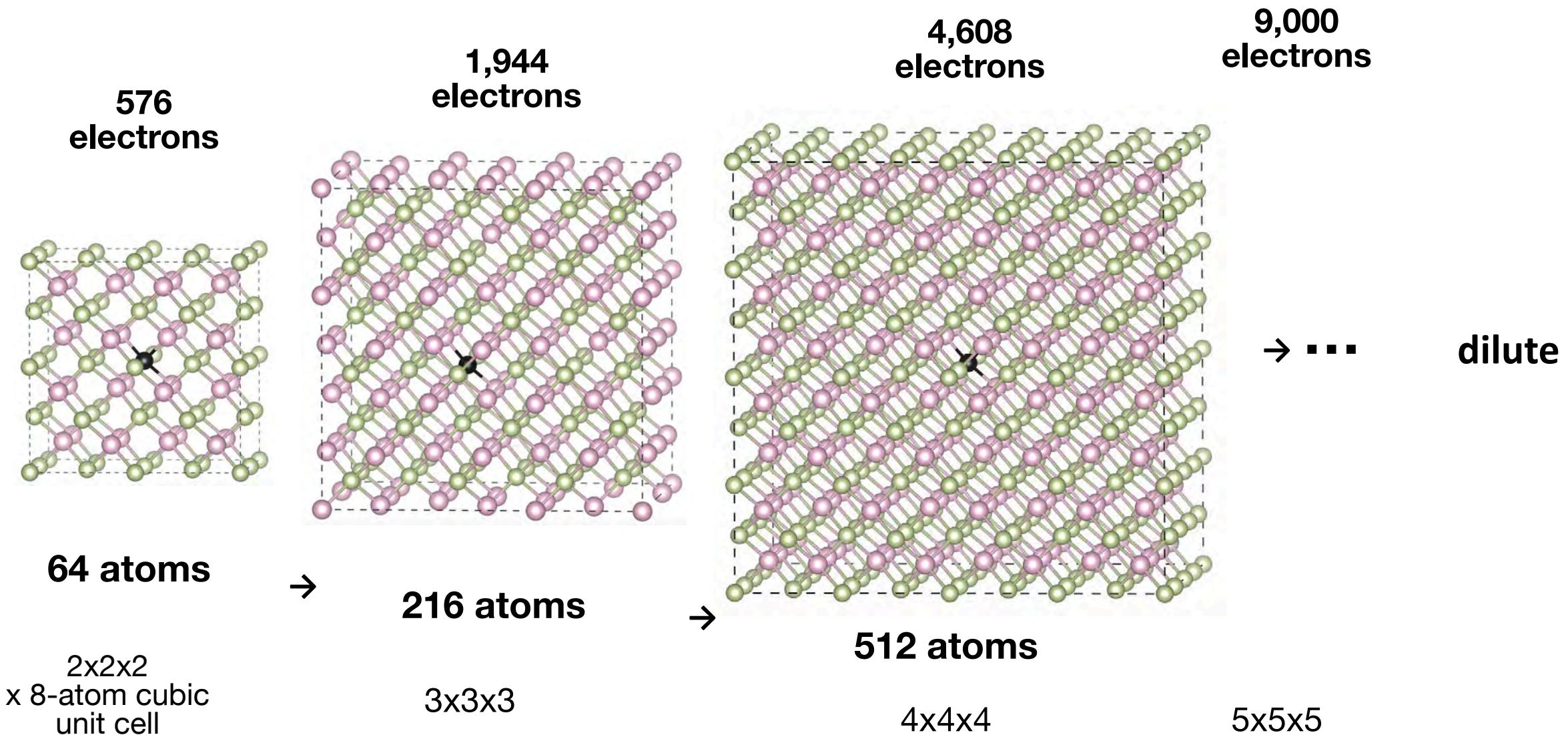
Interactions between defects in neighboring cells



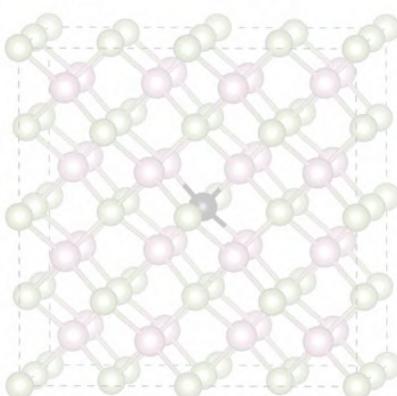
For typical supercell sizes of few 100 atoms

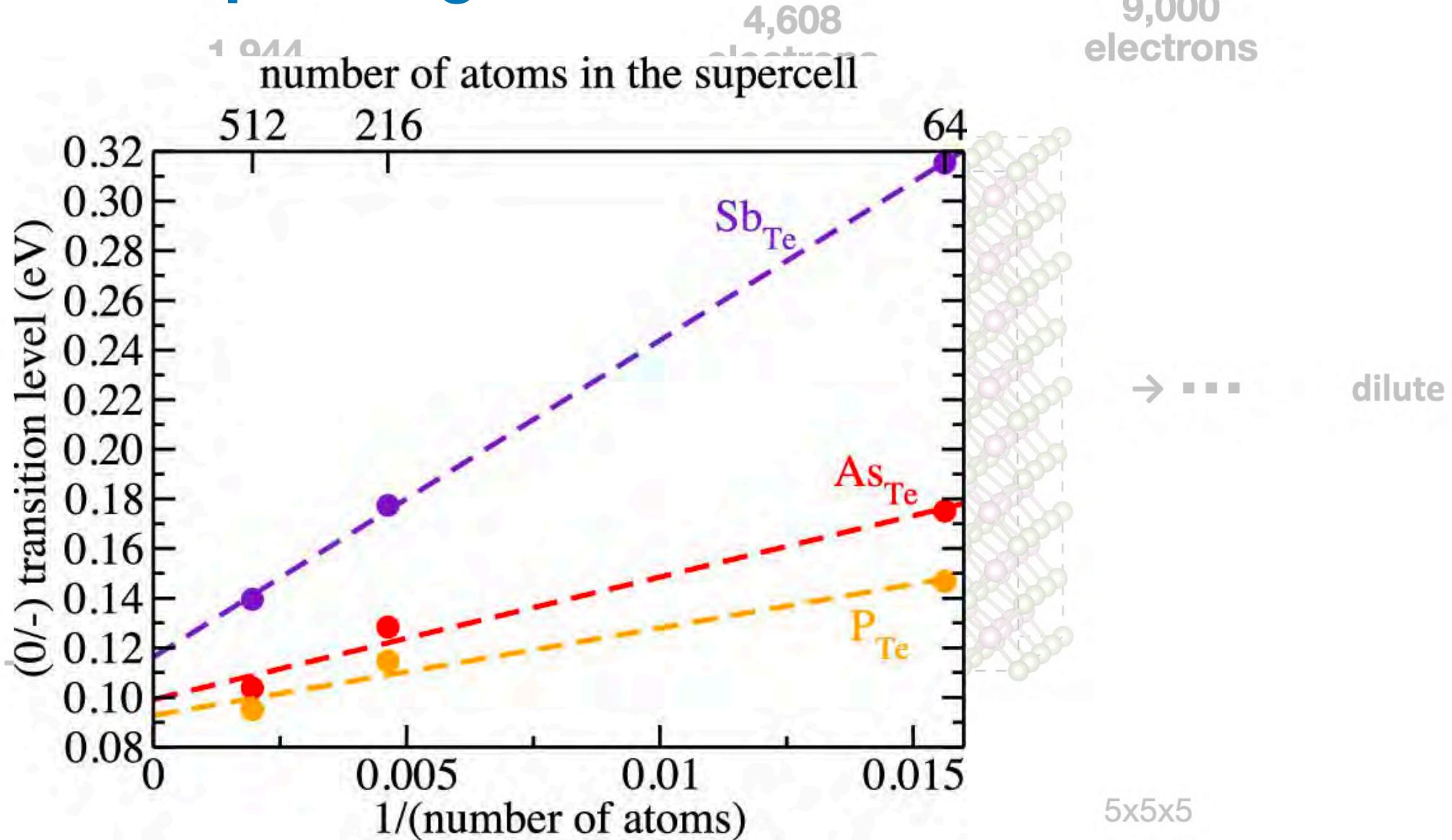
Errors in transition levels (~ 0.1 eV)
are of the order of the transition-level values
for shallow centers

Extrapolating to the dilute limit

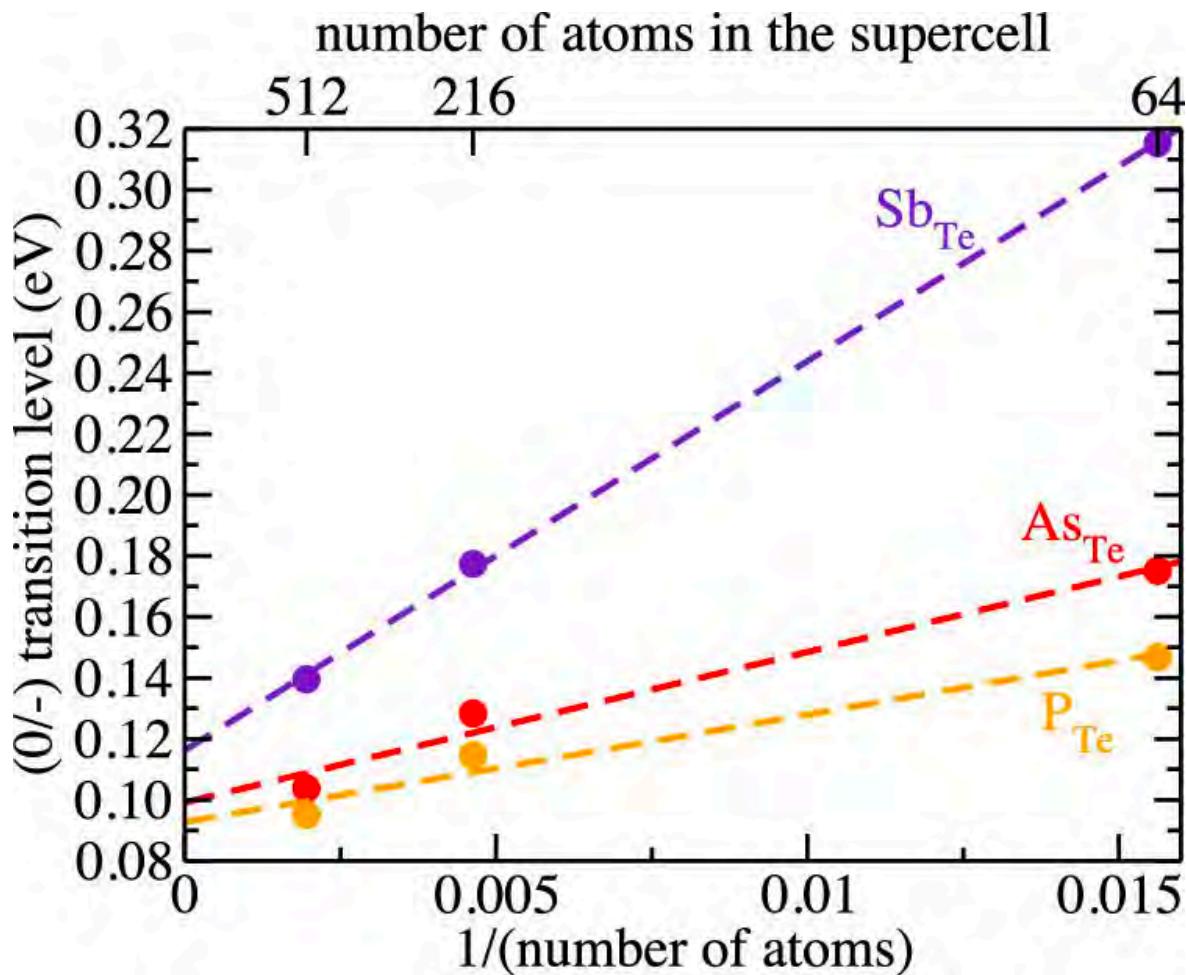


Extrapolating to the dilute limit

576 electrons

64 atoms
2x2x2 repetition of the 8-atom cubic unit cell



Extrapolating to the dilute limit



At dilute limit:

$$P(0/-) = 93 \text{ meV}$$

$$As(0/-) = 99 \text{ meV}$$

$$Sb(0/-) = 116 \text{ meV}$$

Exp.

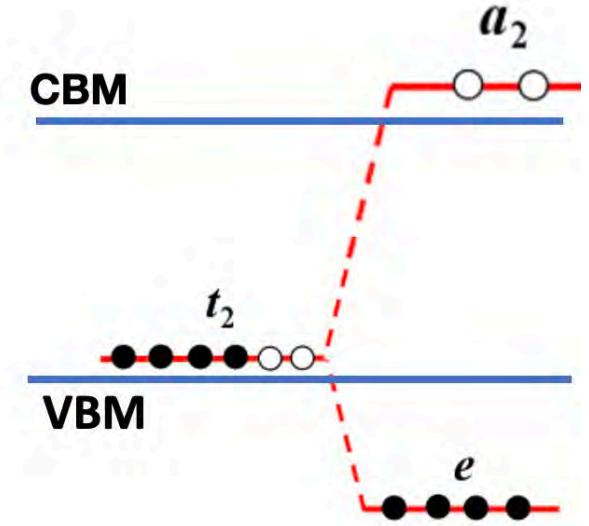
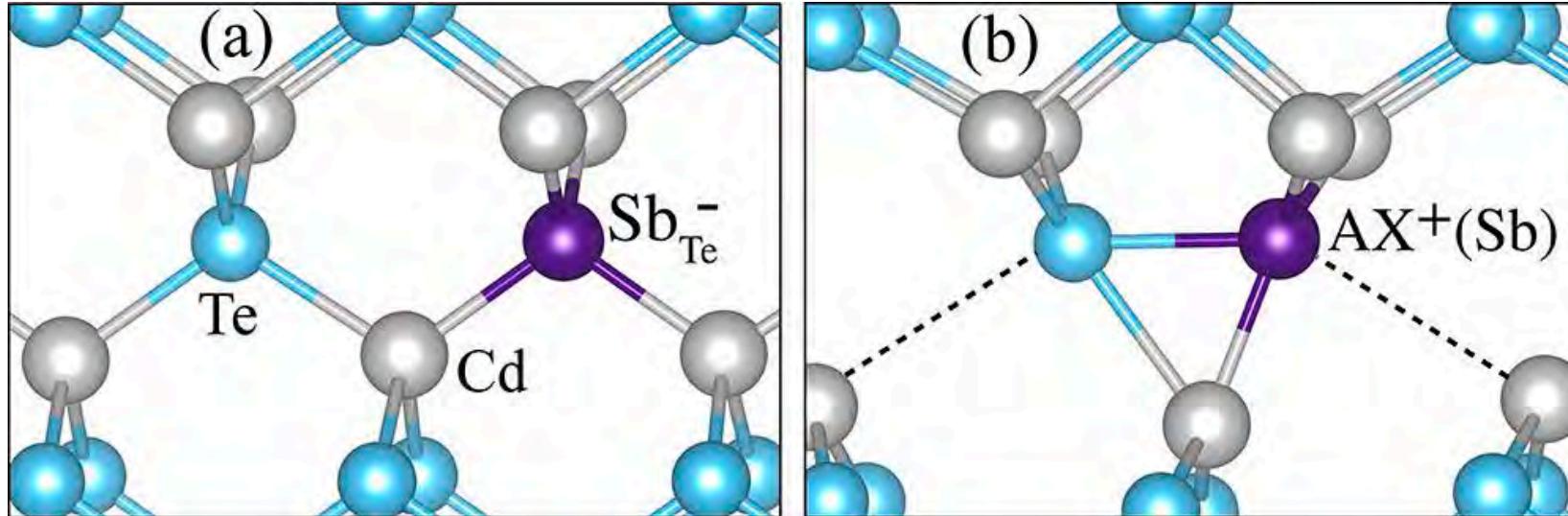
$$P(0/-) = 87 \text{ meV}$$

$$As(0/-) = 94 \text{ meV}$$

$$Sb(0/-) = 103 \text{ meV}$$

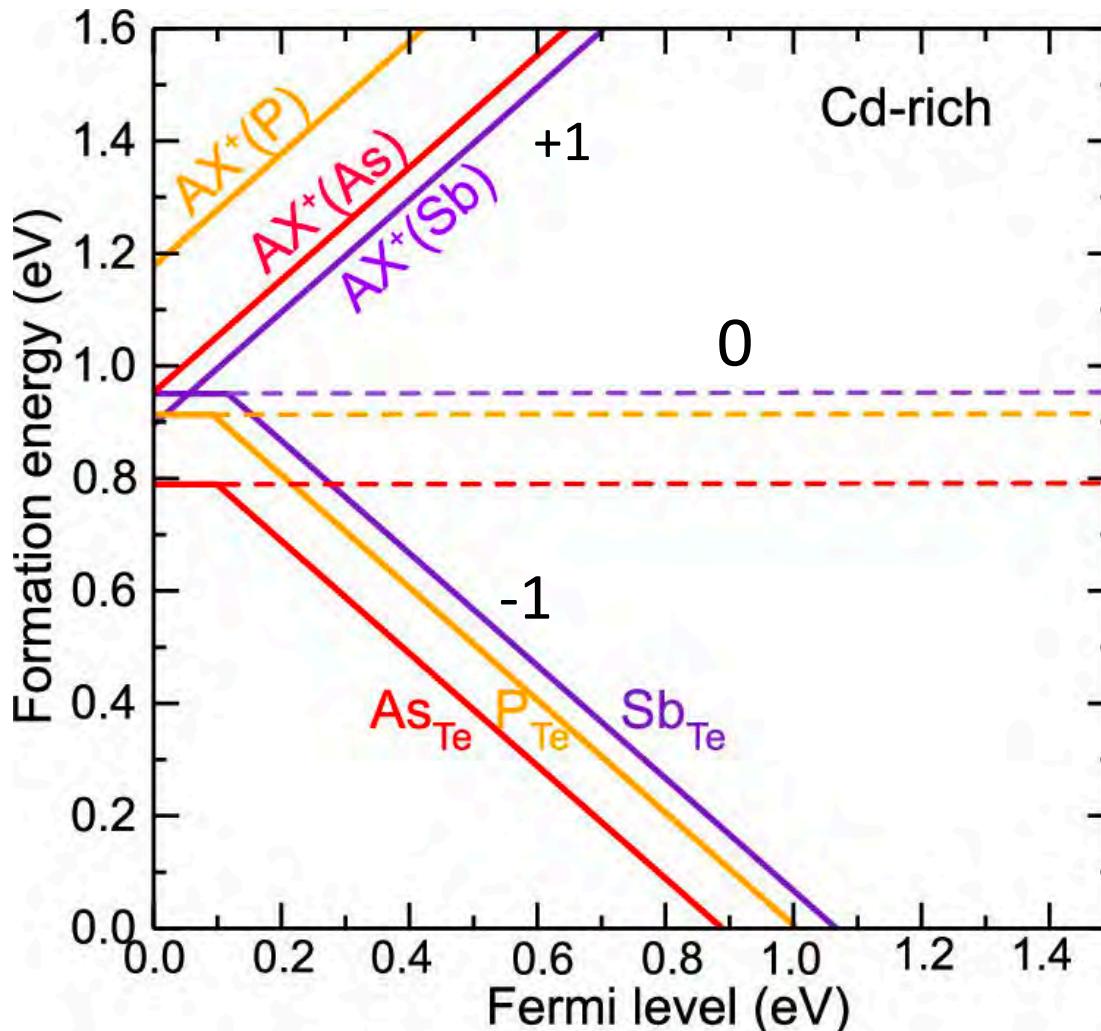
Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Substitutional acceptors vs. AX centers



electronic energy gain
vs.
lattice strain loss

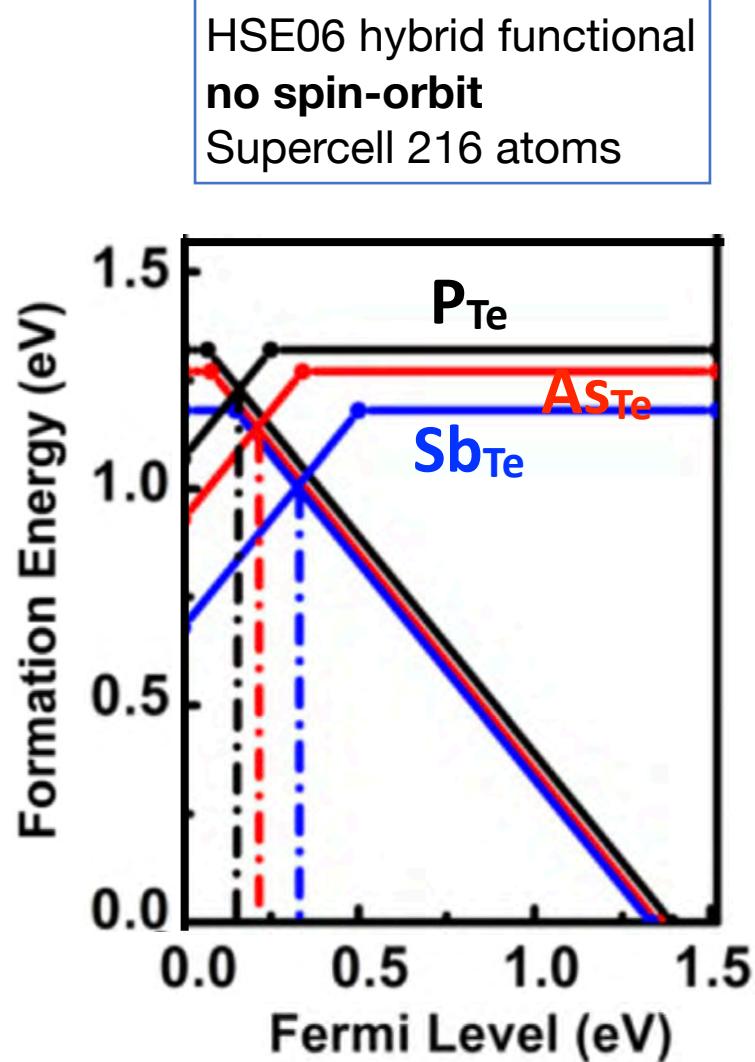
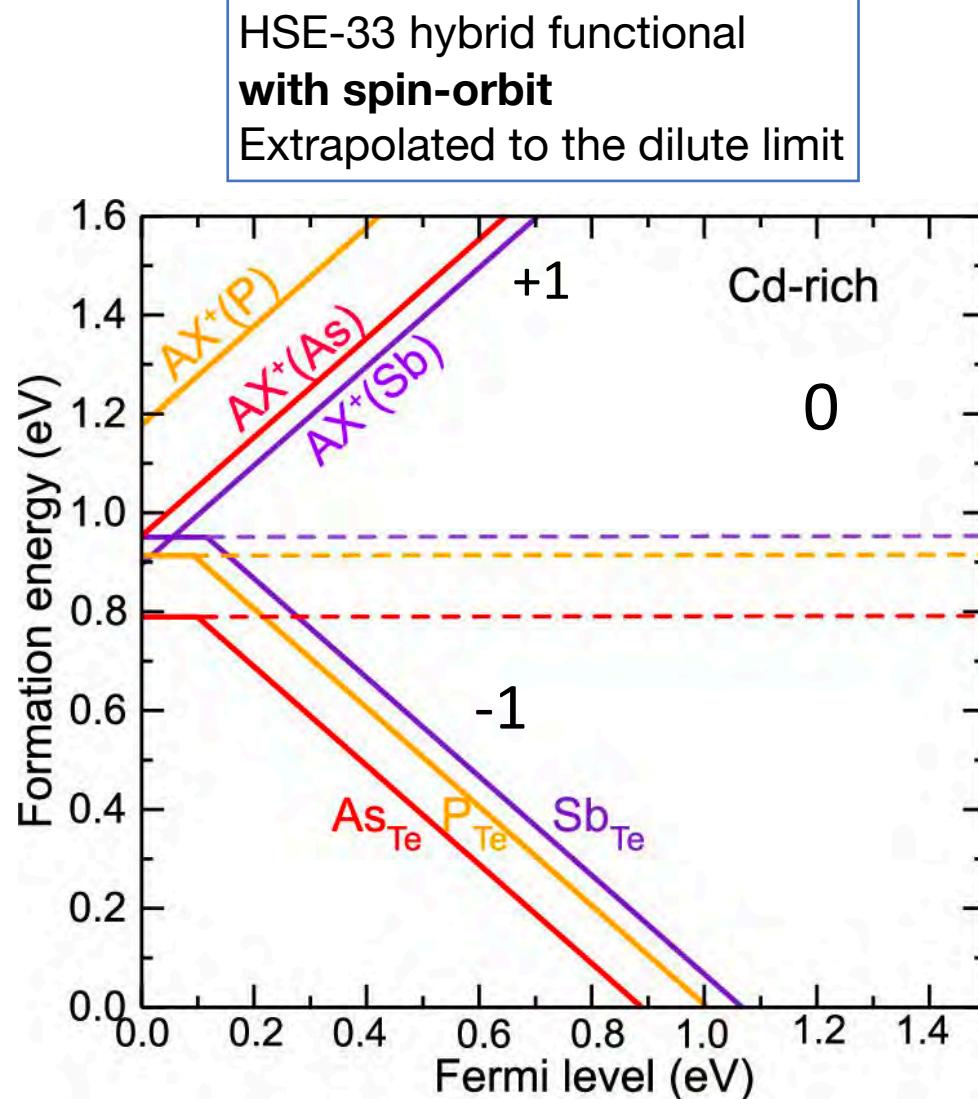
Substitutional acceptors vs. AX centers



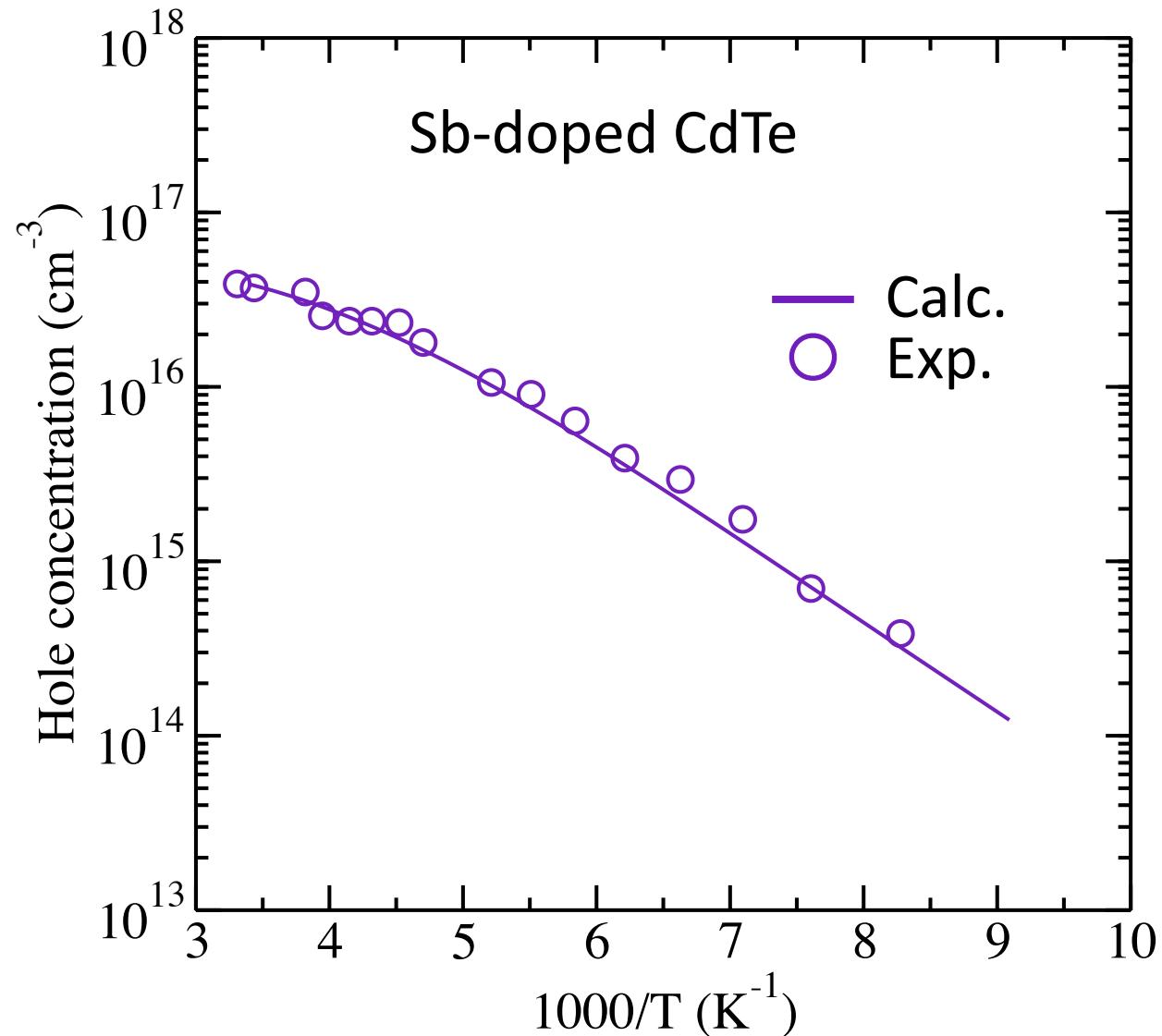
- AX centers are not stable
- P, As, and Te are shallow donors
- (0/-) ionization energies close to expected from hydrogen model

$$13.6 \text{ eV } m^*/\varepsilon^2 \approx 100 \text{ meV}$$

Substitutional acceptors vs. AX centers



Calculated hole concentration for Sb-doped CdTe



Use calc. $E_a = 116 \text{ meV}$

Fit: $[\text{Sb}] = [N_a] = 0.58 \times 10^{17} \text{ cm}^{-3}$

$$\frac{p(p + N_d)}{N_a - N_d - p} = \frac{N_V}{\beta} e^{-E_a/k_B T}$$

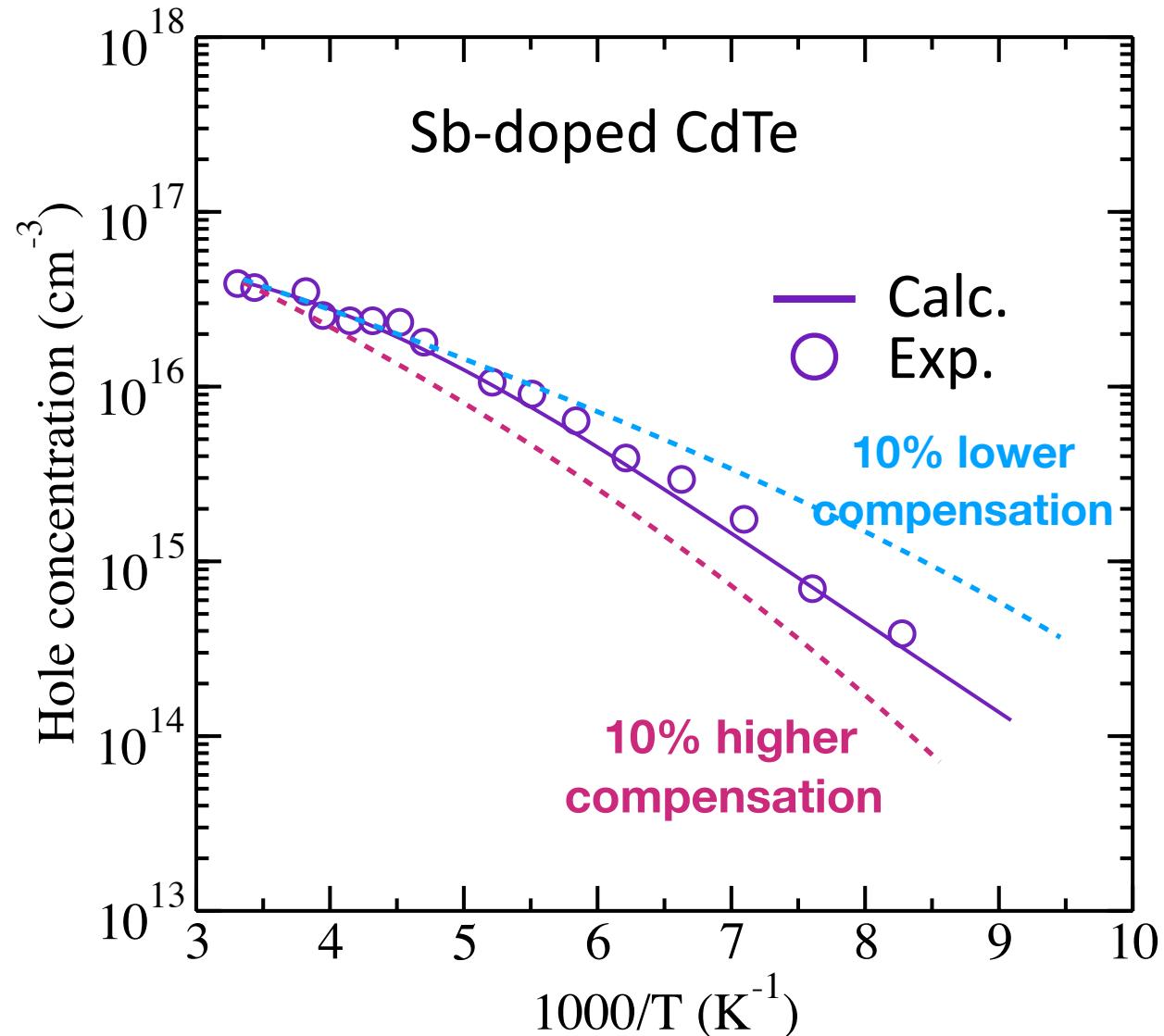
Blakemore, Semiconductor statistics
(Courier Corp., 2002)

N_d are AX centers ~3% of [Sb]

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for Sb-doped CdTe

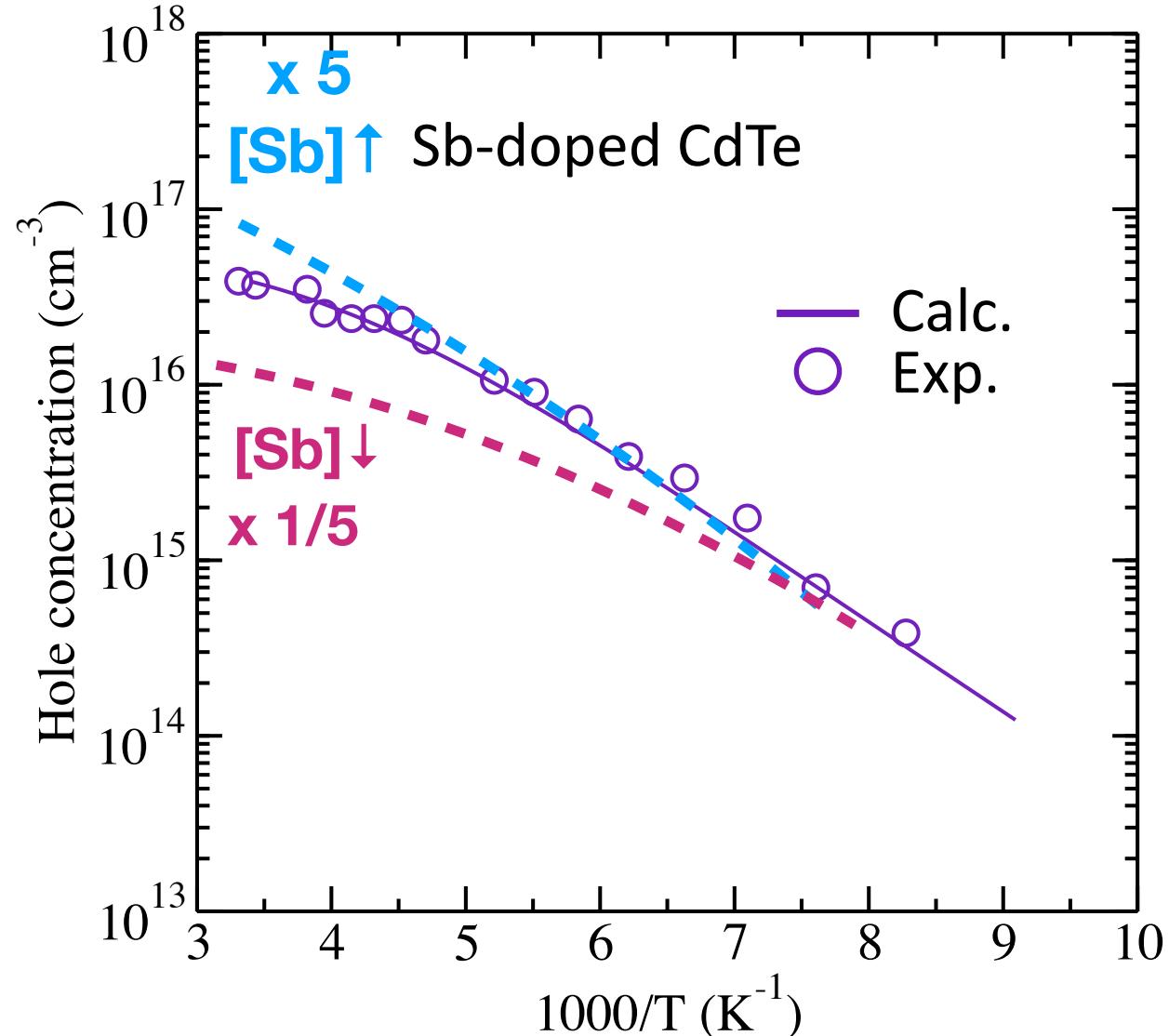


Changing the level of compensation (assuming unknown donor) makes the agreement with experiments worse at low temperatures

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for Sb-doped CdTe

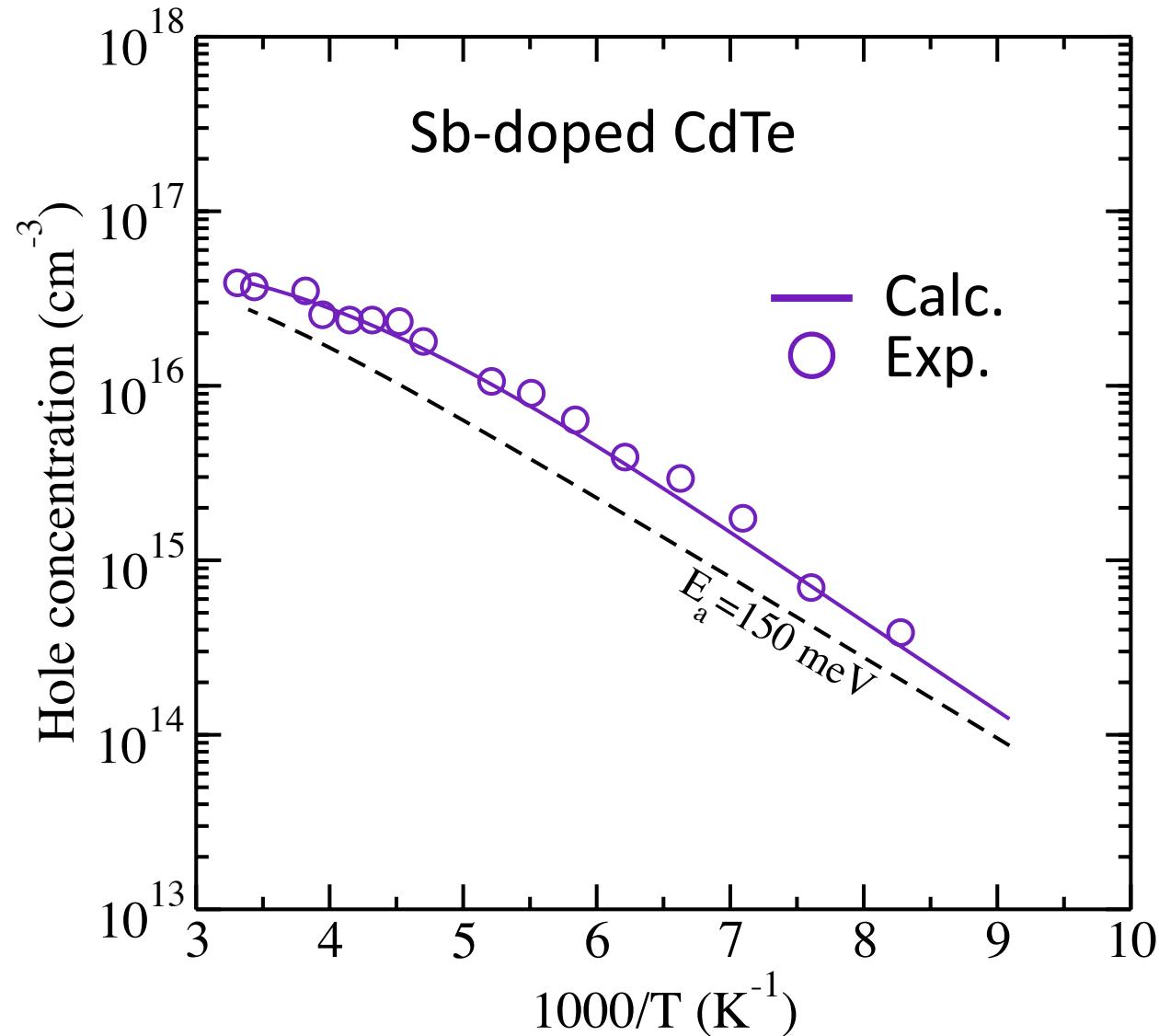


Changing the total Sb concentration by a factor of 5 makes the agreement worse at room temperature

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for Sb-doped CdTe

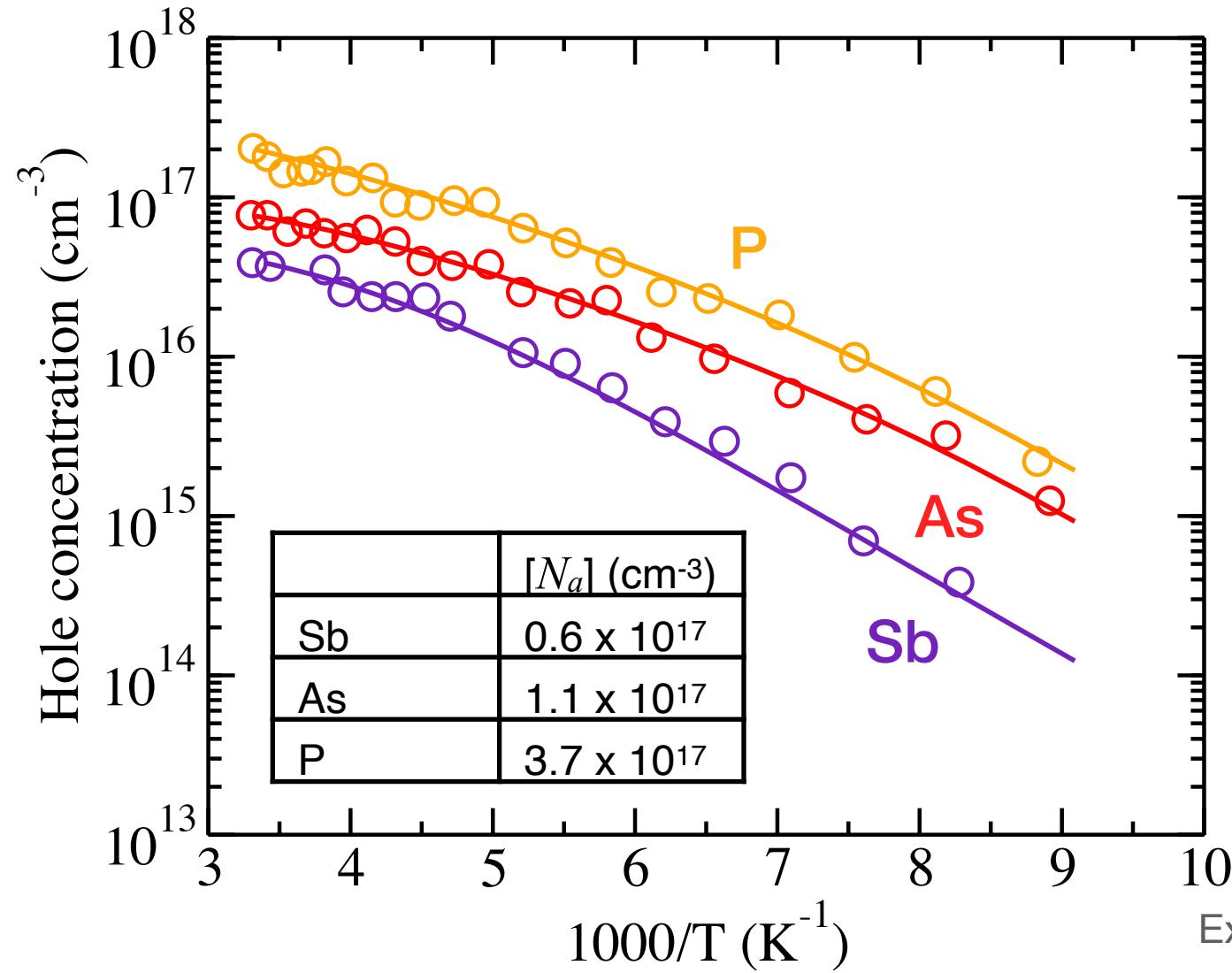


Changing E_a from 116 meV to 150 meV leads to overall lower hole concentration
→ cannot explain experimental data over the whole temperature range

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. **116**, 132102 (2020)

Calculated hole concentration for P, As, and Sb-doped CdTe



Using calc. E_a

P(0/-) = 93 meV

As(0/-) = 99 meV

Sb(0/-) = 116 meV

For P and As, using $[N_d] = 6\% [N_a]$ gives best fit

Source of the compensating donor in the case of P and As still unknown

Exp. data:

Nagaoka *et al.*, Appl. Phys. Lett. 116, 132102 (2020)

Summary

- Need to include spin-orbit coupling and use very large supercell to describe shallow acceptor centers in CdTe
- P, Sb, As in CdTe are shallow acceptors with ionization energies ~ 100 meV, around that of the hydrogen model
- AX center do not play a role as self-compensation center, expect perhaps in the case of Sb under high doping levels
- Best fit to the exp. data of P and As-doped single crystals indicate presence of compensating donors with 6% of the dopant concentration
- Doping efficiency in single crystals decreases at higher doping ($>1E17$ cm $^{-3}$), the cause of which is still unknown
- Low doping efficiency in thin films likely to have contribution from grain boundaries that serve as source or sink of compensating defects.