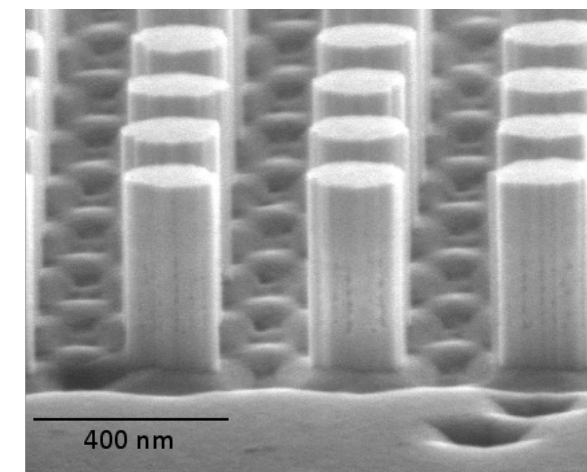
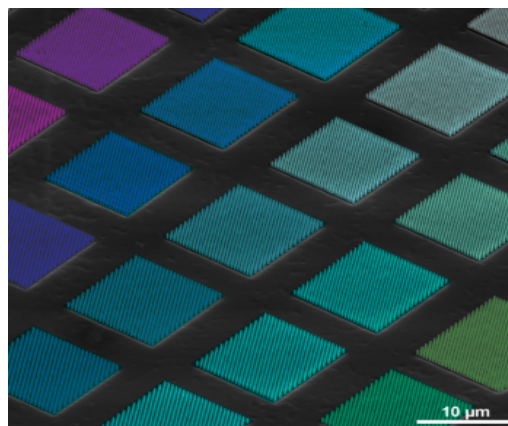
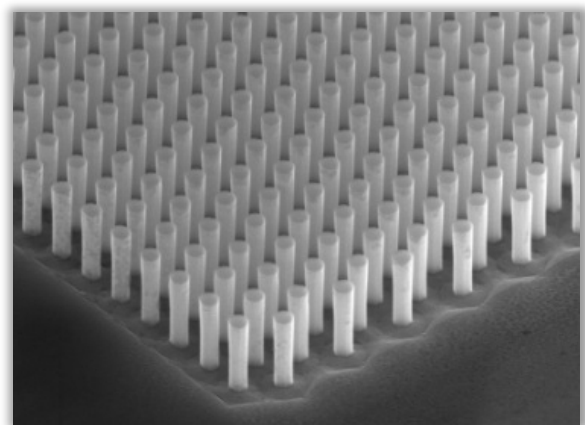


III-Nitride Nanophotonics

Ganapathi (Ganesh) Subramania

Semiconductor Materials and Device Sciences,
Sandia National Laboratories, Albuquerque, NM

OSE Seminar
UNM, December 9, 2021



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Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

Ubiquitous presence of III-nitride based solid state lighting has opened up III-nitrides as important materials for visible and UV nanophotonics. Nanophotonic architectures such as photonic crystals and metamaterials/metasurfaces have become key players in modern photonics. They have enabled unprecedented capabilities combined with great versatility to control various properties of light -propagation, polarization, emission and photon statistics. Unique materials processing challenges posed by III-nitride materials motivates us to explore alternate techniques for nanostructure fabrication compared to Si and III-Vs. In this talk, I will present some of the approaches and results of III-N nanophotonics research based on photonic crystals and metamaterials performed at Sandia.

Collaborators

Sandia

- Nick Karl
- Keshab Sapkota
- George Wang
- Igal Brener
- Art Fischer
- Dan Koleske
- Duke Anderson
- Jeremy Wright
- Jeff Tsao

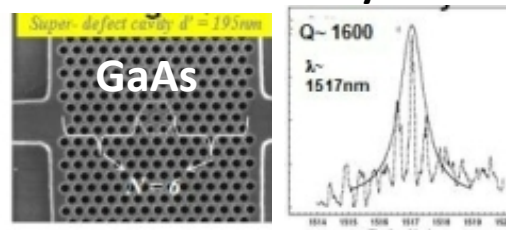
UNM

- Daniel Feezell
- Zachary Meinelt

Nanophotonics Activity

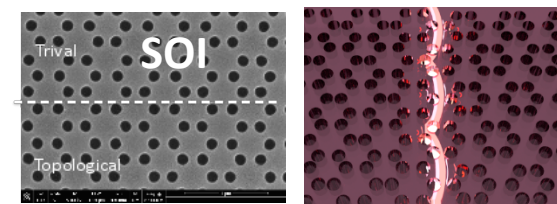
Integrated Photonics, Nanoscale lasing, Emission control, Solid State lighting, Energy conversion, Non-classical light sources, Subwavelength light control, Detection, Sensing , Nanocircuitry

2D PhC cavity



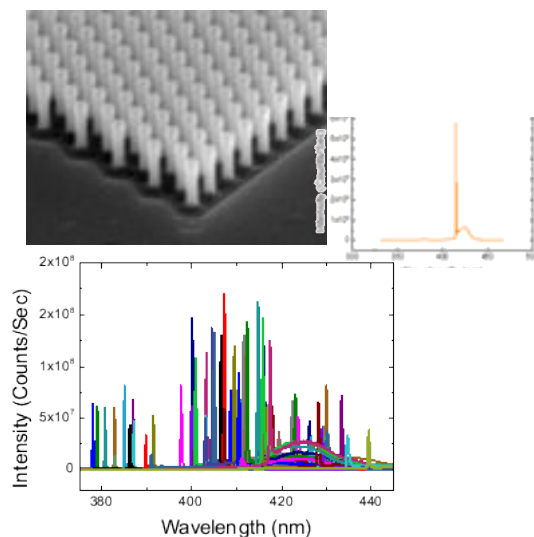
Subramania, G., et. al., *J. M. App. Phys. Lett.* **83**, 4491(2003).

Topological Photonics



P. D. Anderson and G.Subramania, *Optics Express* **25** , 23293 (2017).

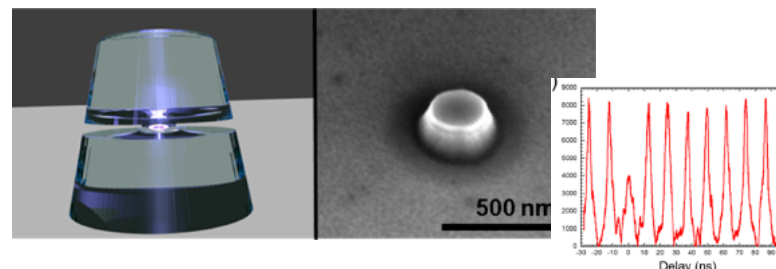
GaN PhC multicolor lasers



Wright, J. B. *et al. Sci. Rep.* **3**, 2982(2013)

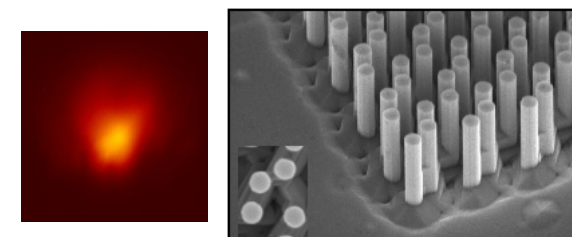
2D Photonic Nanostructures

GaN QD for single photon sources



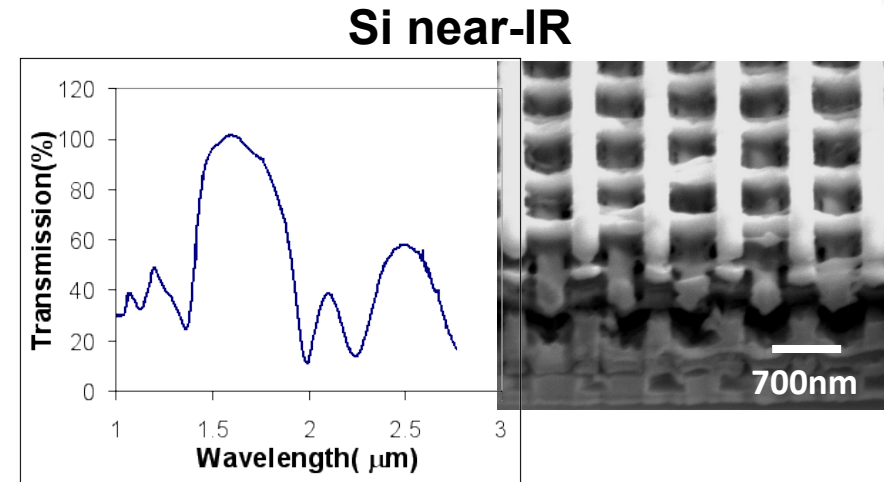
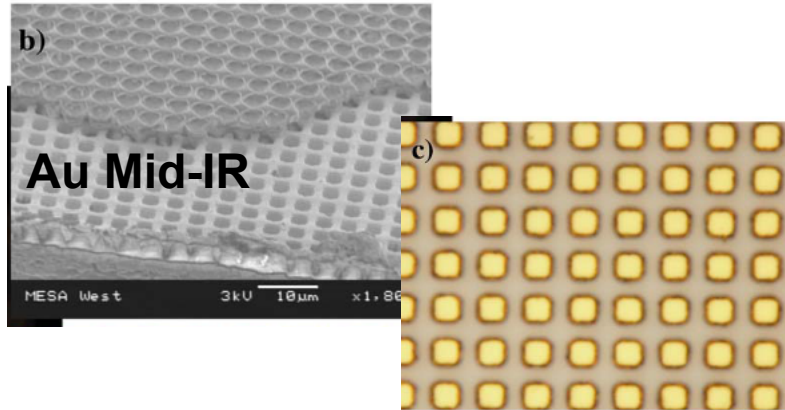
Fischer, A. J. et. al, . *ACS Phot.* **4**, 2165, (2017).

GaN quasi-periodic



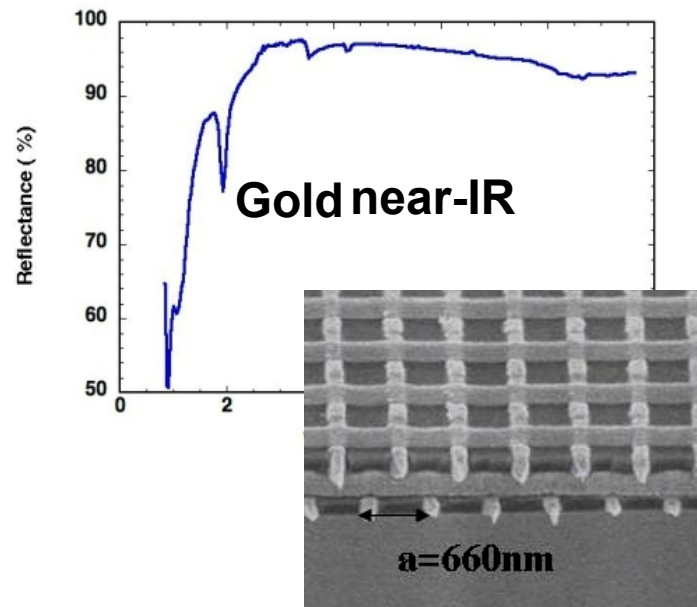
Anderson, P. D. et. al.. *Opt. Mat. Exp.* **7**, 3634-3642, (2017).

Nanophotonics Activity

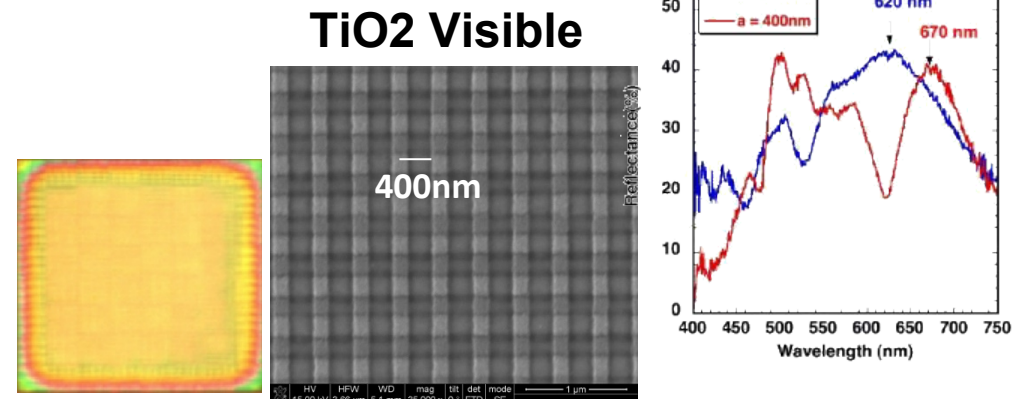


G. Subramania and S. Y. Lin, Applied Physics Letters 85 (21), 5037 (2004)..

3D Photonic Crystals

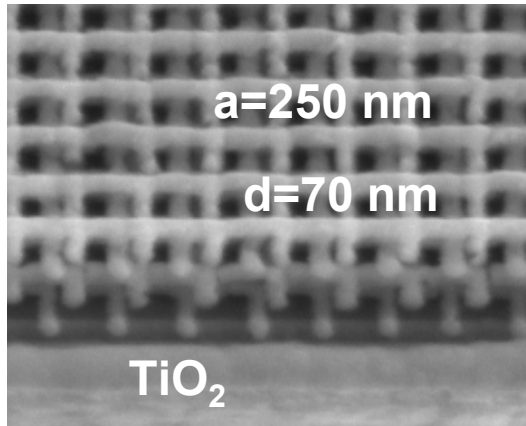


Subramania, G.; Rivera, J. M., Organic and Nanocomposite Optical Materials 2005 MRS proc **846** 321

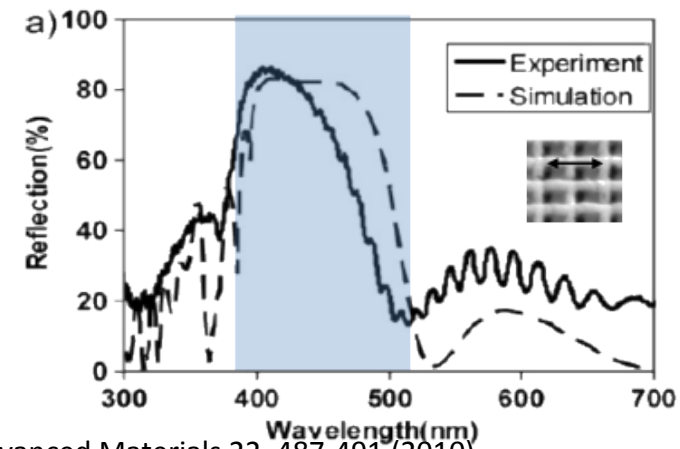


G. Subramania, Y-J. Lee, I. Brener, T.S. Luk, and P.G. Clem, Optics Express 15 (20), 13049 (2007).

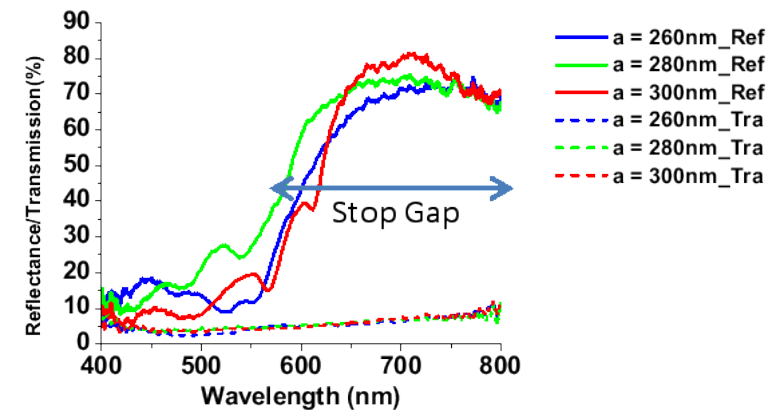
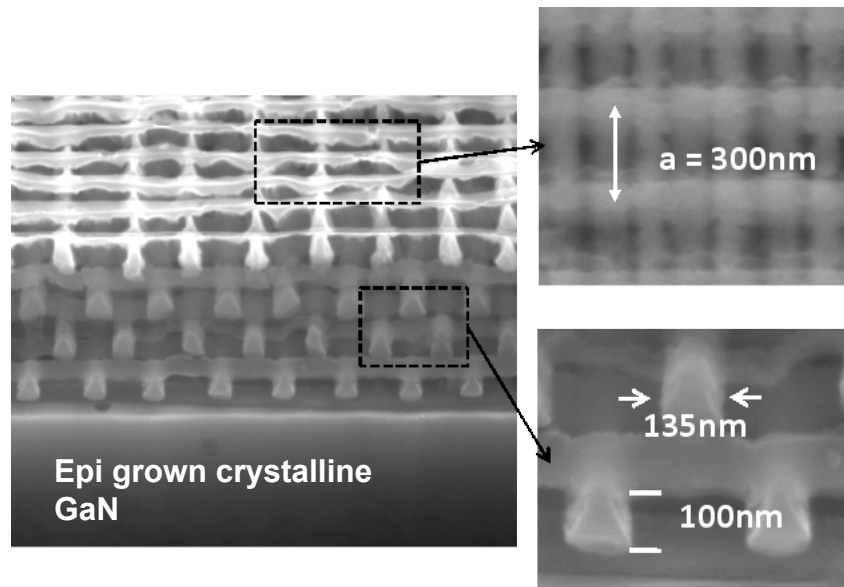
Nanophotonics Activity



G. Subramania, Y.-J. Lee, A. J. Fischer, and D. D. Koleske, Advanced Materials 22, 487-491 (2010)



3D Photonic Crystals



G. Subramania, Q. Mi Li, Y.-J. Lee, J. J. Figiel, G. T. Wang, and A. J. Fischer, Nano Lett. 11 4591 (2011).

Outline

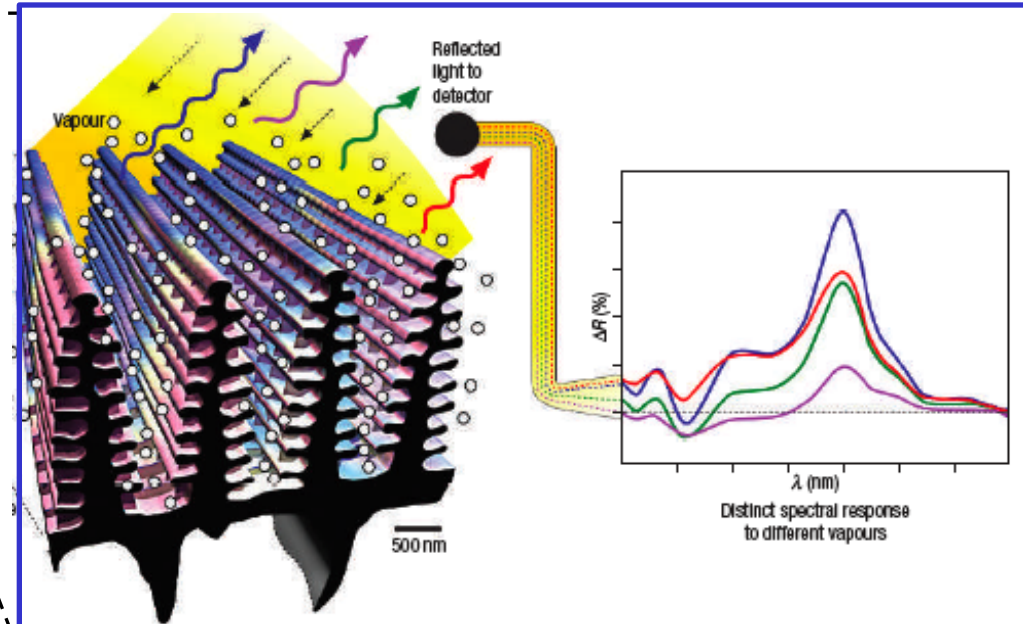
- Introduction on light control with nanostructuring
- Lasing in III-N Nanowire 2D Photonic Crystals
 - Photonic band structure, fabrication, emission profile
- III-Nitrides for Metasurfaces
 - Forward scattered transmission, cladding for substrate isolation
- Quantum Dots for Single Photon Sources (time permitting)

Controlling Light via Nanostructuring : in Nature

- Color response highly sensitive to different vapors

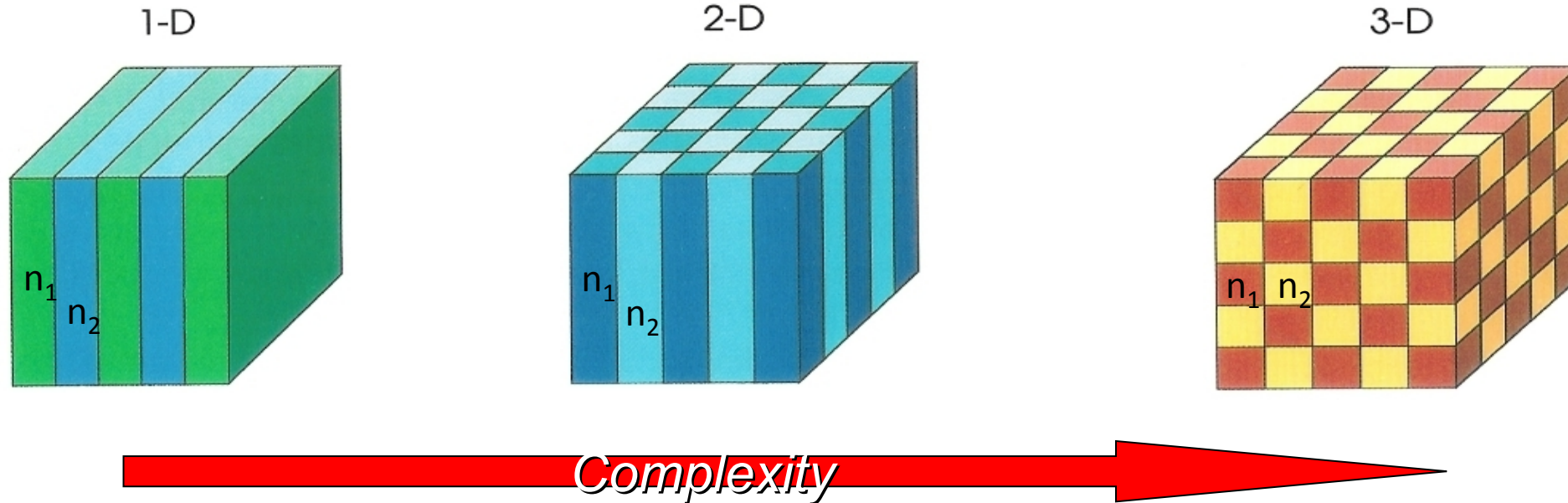


Morpho Butterfly
(M. Sulkowskyi)



* R.A. Potyrailo et. al. , Nature Photonics , 1, 123(2007).

Photonic Crystals: Nanostructured Electromagnetic Environment



Control of light propagation

- Guiding, bending and splitting
- Negative refraction
- Self-collimation
- Localization
- Slow light
- Cloaking

Control of light emission

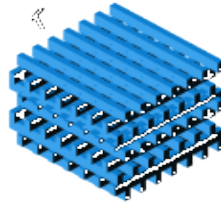
- Spontaneous emission enhancement
- Strong-coupling (Photon – Atom Bound states, Non-Markovian emission process)
- Non-classical light sources

Reference: “Photonic crystals: Molding the flow of light”, J.D. Joannopoulos, R.D. Meade, J. Winn, Princeton Univ.Press, NJ(1995)

Spontaneous Emission Control using PC

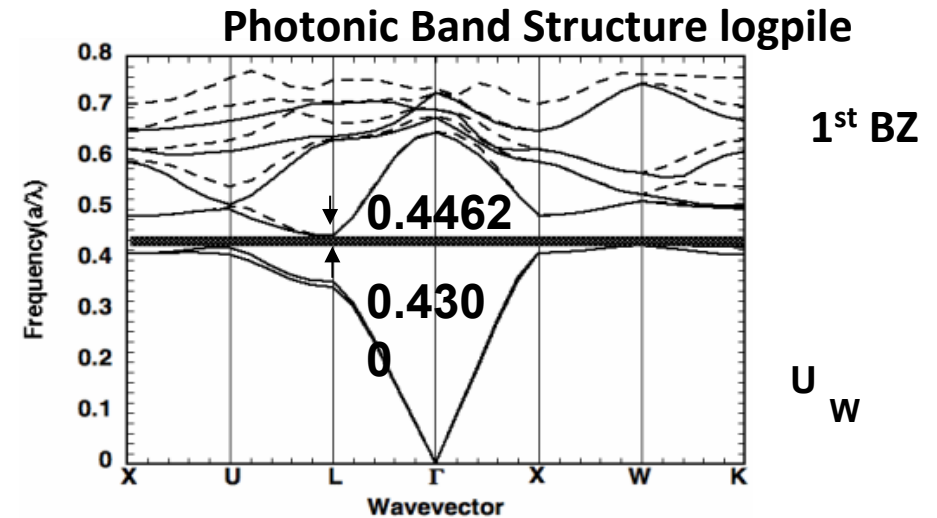
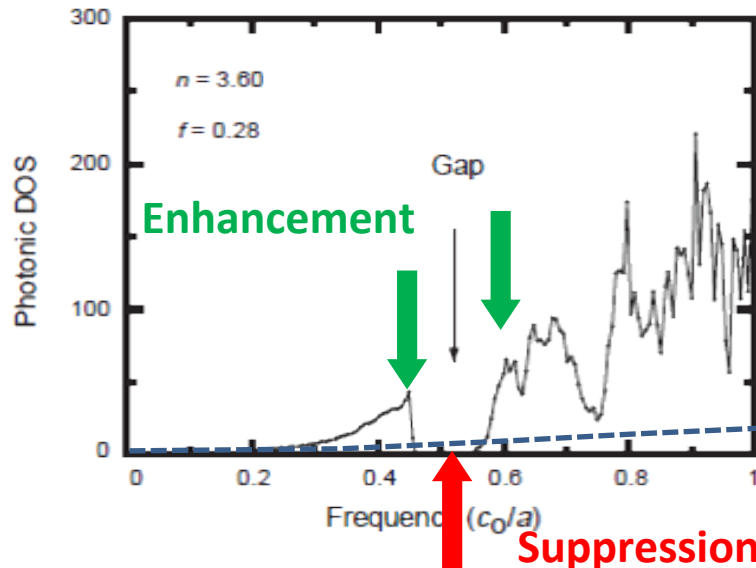
Radiative rate of an emitting dipole in an EM field (Fermi's Golden Rule)

$$\Gamma_r(\omega) = \frac{\pi \omega \mu^2}{3 \epsilon_0 \hbar} \rho(\omega)$$



Photonic Density of States (PDOS)

$$\rho_f(\omega) = \frac{\omega^2}{\pi^2 c^3} \leftarrow \text{in free space}$$



Radiative rate modification

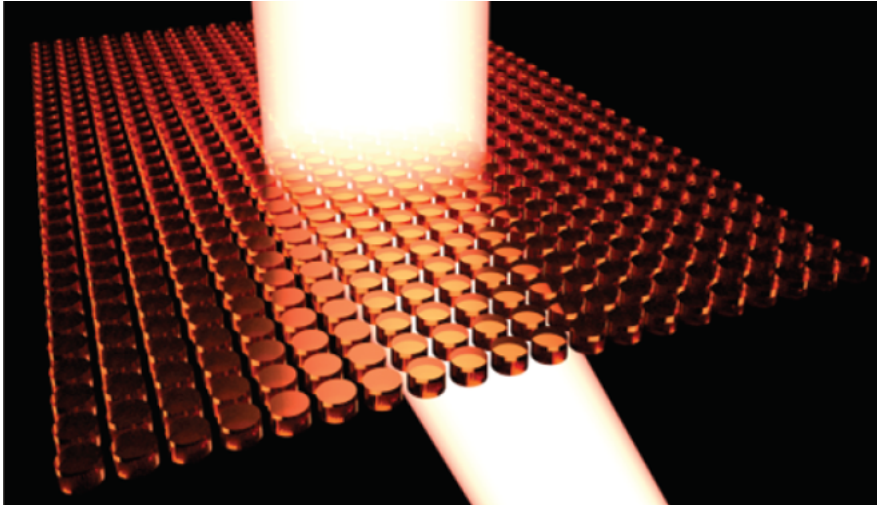
$$\frac{\Gamma_{r_PC}(\omega)}{\Gamma_{r_free}(\omega)} = \frac{\rho_{PC}(\omega)}{\rho_f(\omega)}$$

Local Photonic Density of States (LDOS)

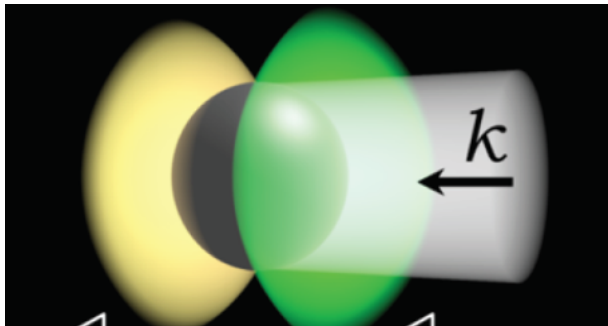
$$\rho_L(\mathbf{r}, \omega) = \sum_n \int_{1stBZ} d^3k |\bar{\boldsymbol{\mu}} \cdot \mathbf{E}_{n,k}(\mathbf{r})|^2 \delta(\omega^2 - \omega_{n,k}^2)$$

Huygen's Metasurfaces

Scatterers perpendicular to light propagation

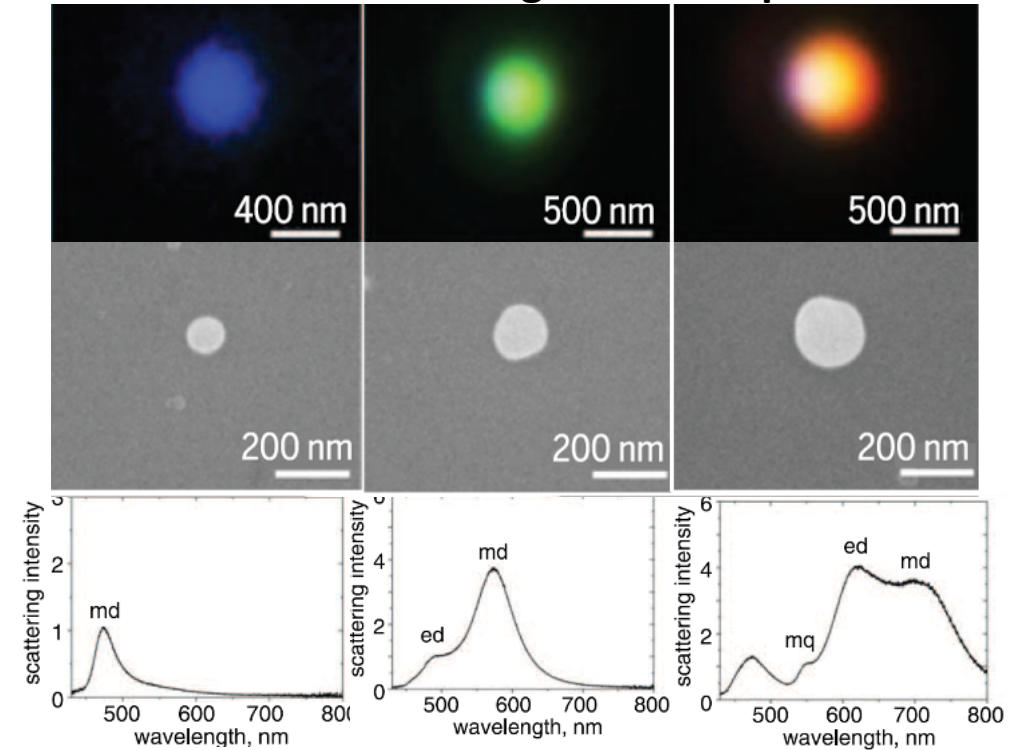


High efficiency forward scattering occurs when ED and MD align

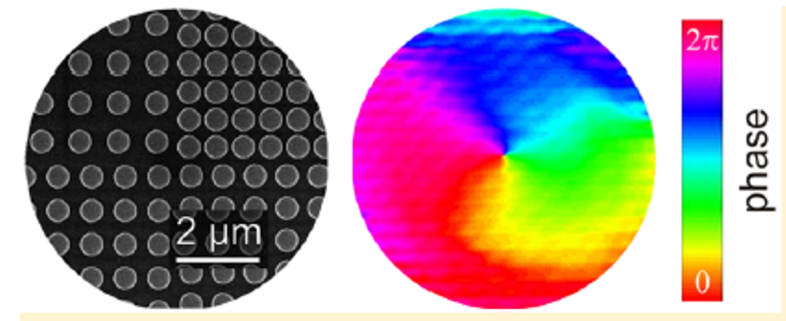


A.Kuznetsov et. al. Science **354** (6314), aag2472 (2016).

Electric and Magnetic Multipoles



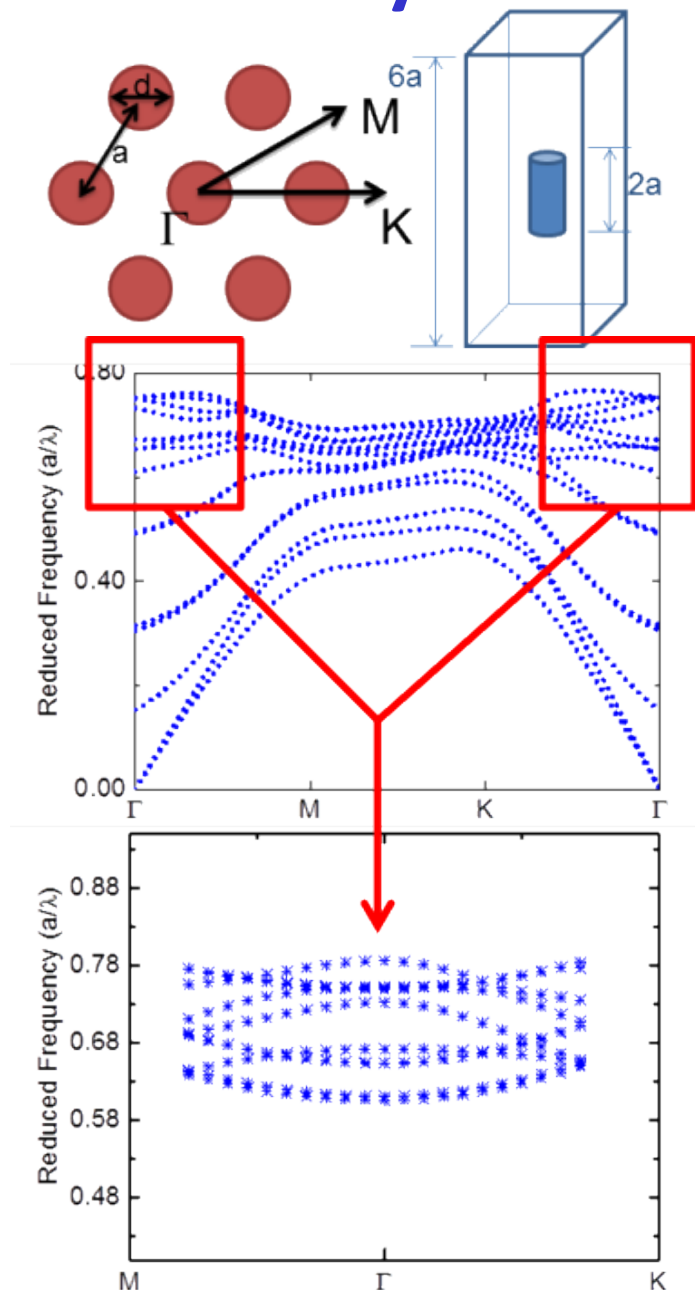
Optical Wavefront Control



Katie E. Chong et. al. Nano Lett. **15** (8), 5369 (2015)

Lasing in III-N Nanowire 2D Photonic Crystals

Vertically emitting photonic crystal lasers - Theory

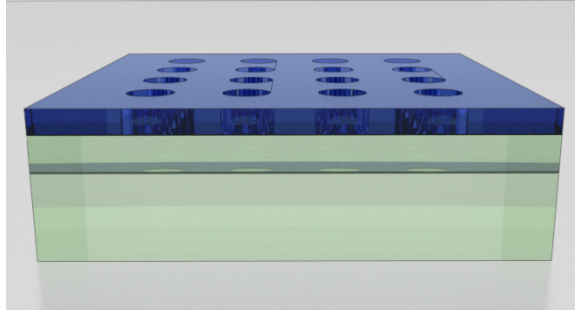


- *Low lasing threshold*
 - *Increase interaction of light with the gain region*
 - *Enabled by 'Low group velocity modes' in photonic crystal**
- *Vertical Emission*
 - *At higher bands there are low group velocity modes near the Gamma point*
 - *Near the Gamma point the in-plane wave vectors are small resulting in near-normal emission*
- *Wavelength Tunability*
 - *Achieved through lattice constant and nanowire diameter variation*

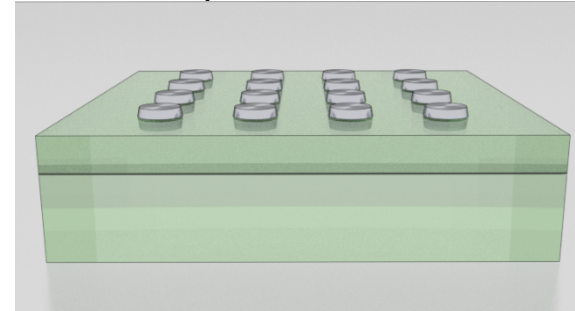
*Sakoda, K. *Optical Properties of Photonic Crystals*. (Springer-Verlag, 2001).

Fabrication of III-N nanowire laser array

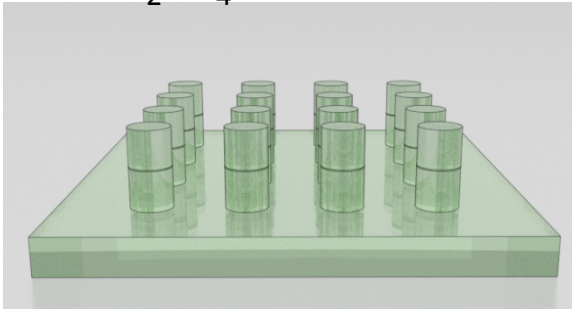
EBL pattern in PMMA



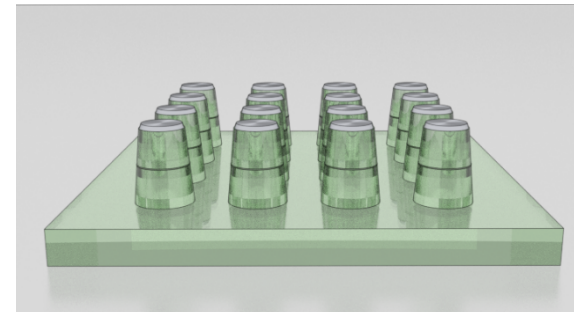
Ni evaporation and lift-off



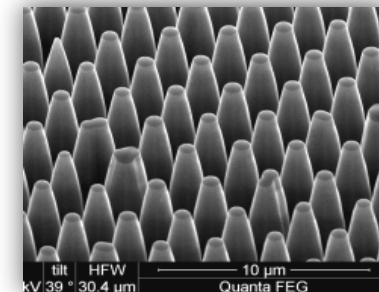
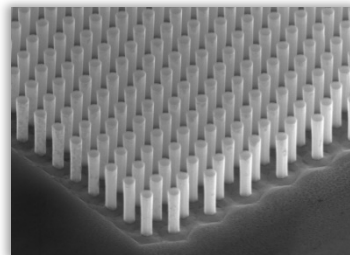
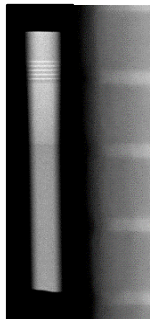
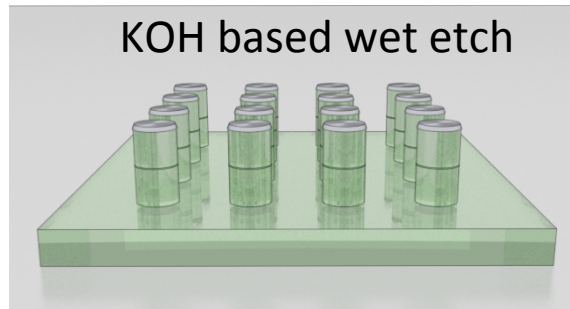
H₂SO₄-based Ni removal



Cl₂ based dry etch

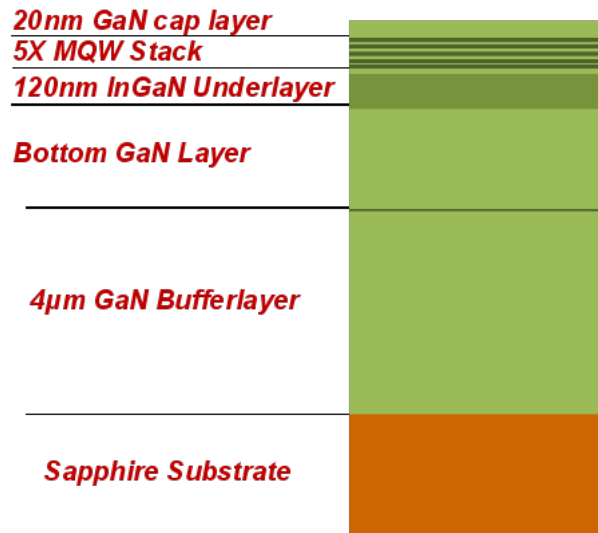


KOH based wet etch

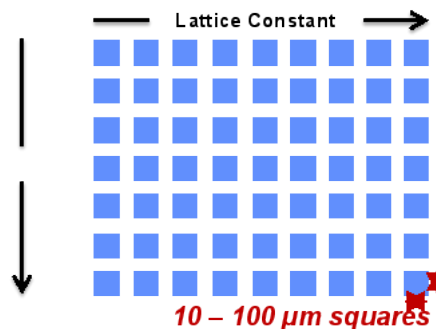


Fabrication of III-N nanowire laser array

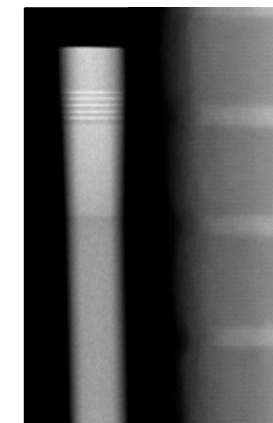
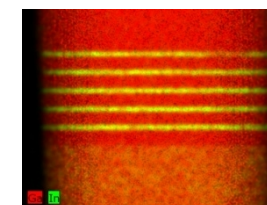
Growth of III-N active material



Patterning of 2DPC using e-beam lithography

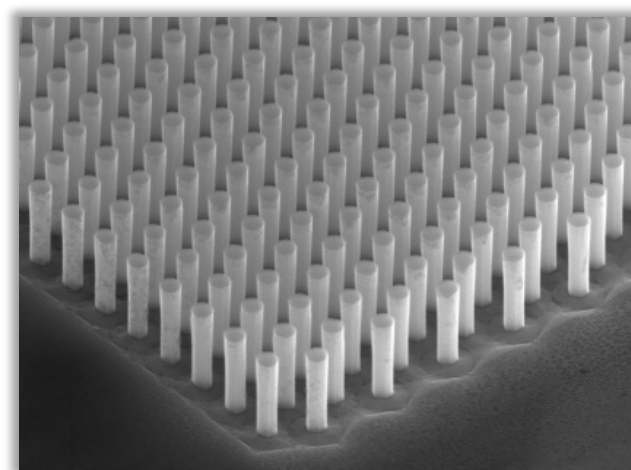
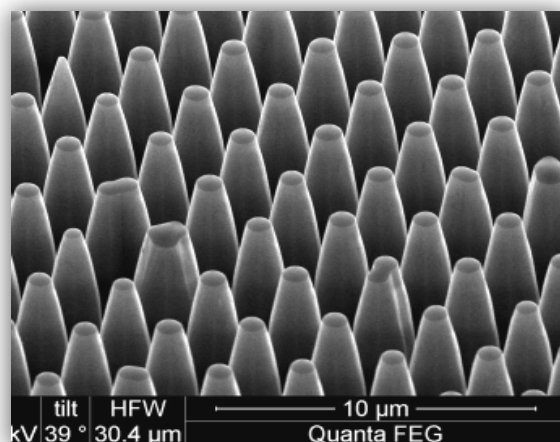


TEM images



Ping Lu,
Sandia

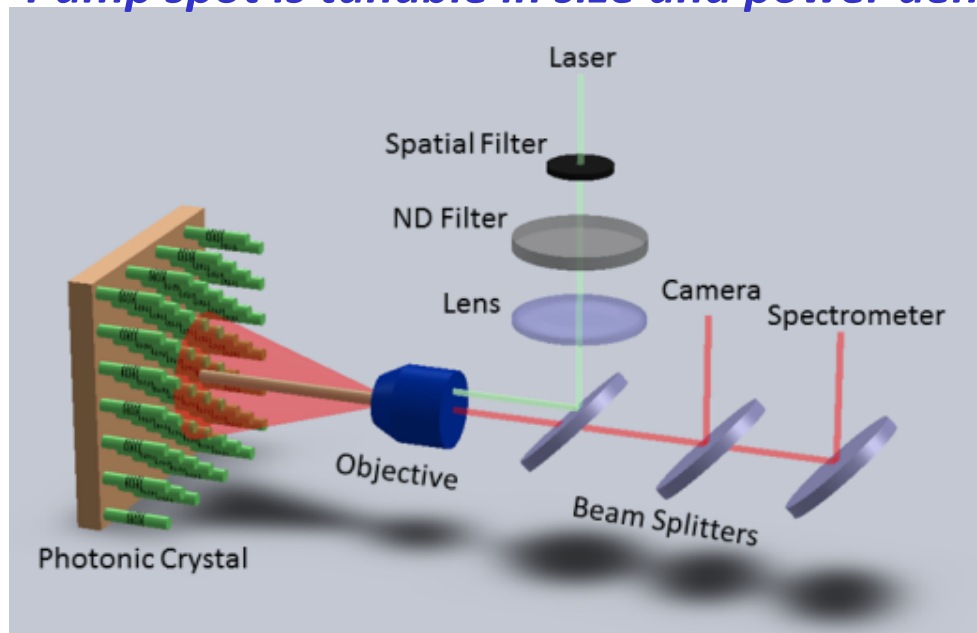
Inductively coupled plasma dry etch



Wet etch using KOH process

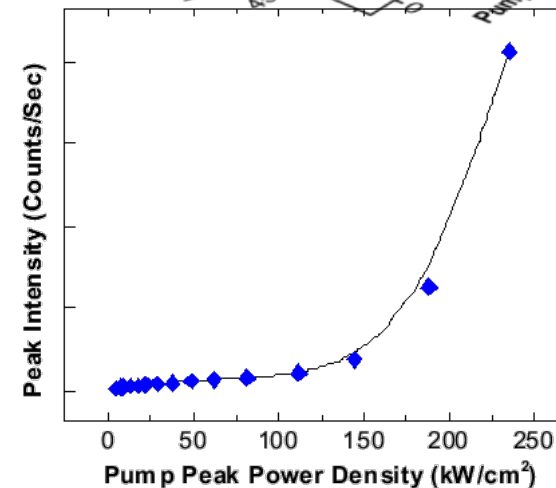
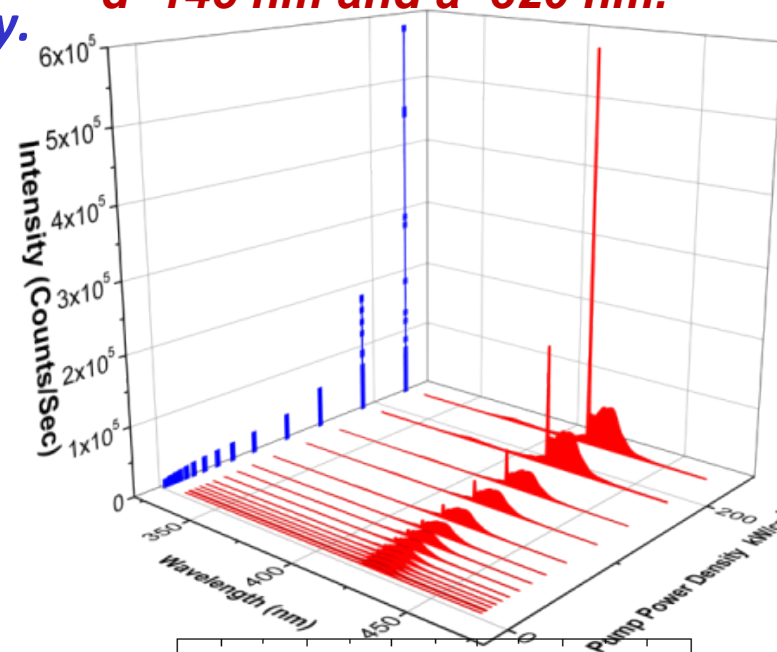
Optical response – low threshold lasing

Pump spot is tunable in size and power density.

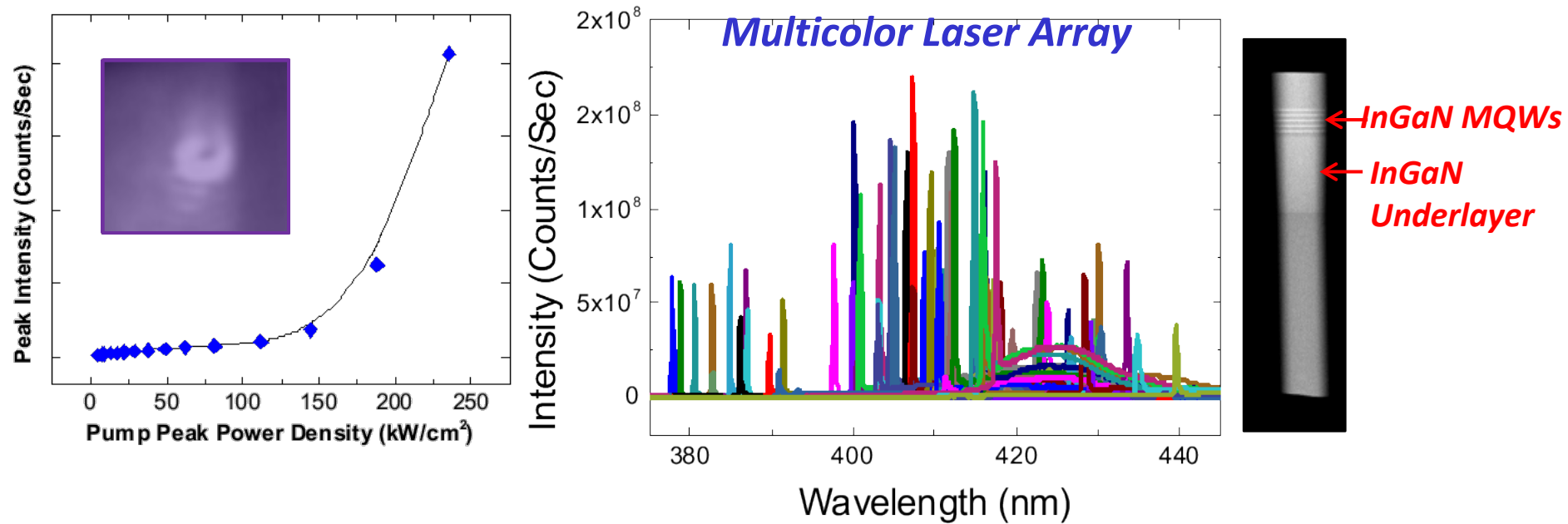


A low lasing threshold is achieved with $< 500 \text{ kW/cm}^2$ for all PC lasers fabricated.

**Array Element Dimensions:
 $d=145 \text{ nm}$ and $a=320 \text{ nm}$.**



Optical response – low threshold lasing



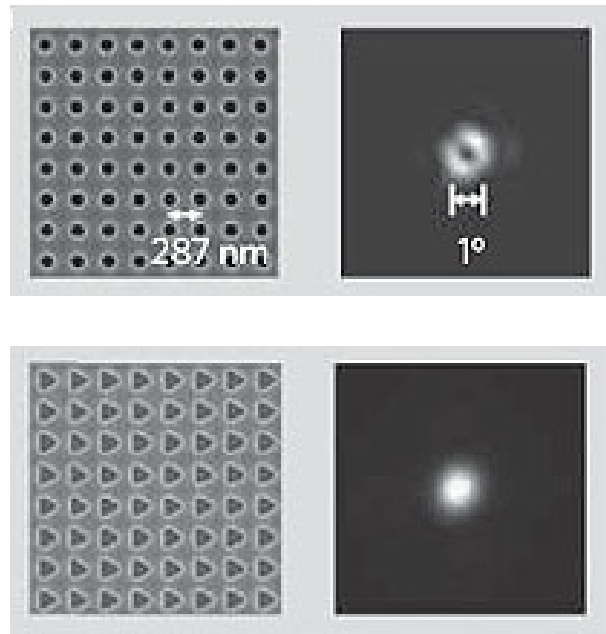
- A low lasing threshold is achieved with $< 500 \text{ kW}/\text{cm}^2$ for all PC lasers fabricated.
- Wavelength tunability (lattice constant and nanowire diameter).
- Multiple color laser emission on a single epitaxial wafer covering a spectral range from 380-440nm.
- Multiple gain regions can simultaneously lase
- Pathway for enhancing the available lasing spectral range

Wright, J. B. *et al.* Multi-Colour Nanowire Photonic Crystal Laser Pixels. *Sci. Rep.* **3**, 2982,(2013)

Modifying Extraction and Emission Mode Profile

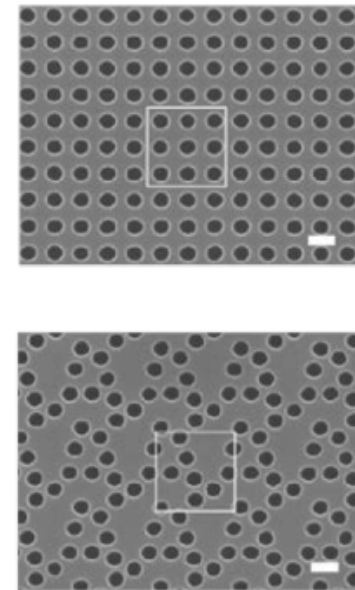
Modify extraction $\rightarrow$ break mirror symmetries

Method 1



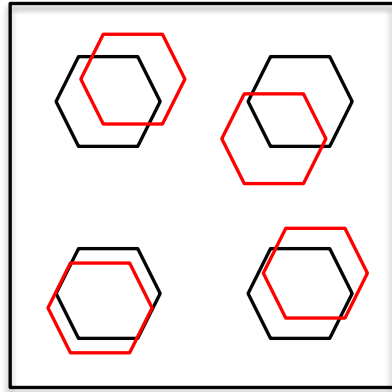
Miyai, E., et al., *Nature*, 441, 946 (2006)
Noda Group

Method 2

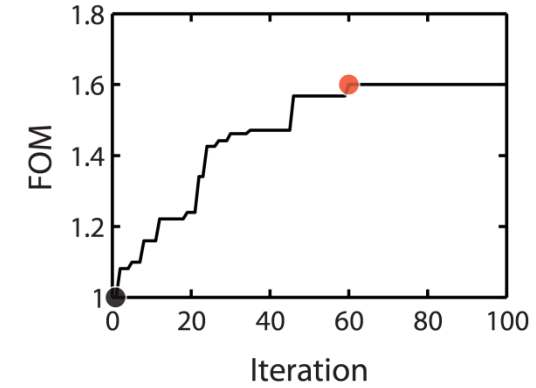


C. Lin, et al., *Optics Express*, 21, A872 (2013)
Povinelli Group

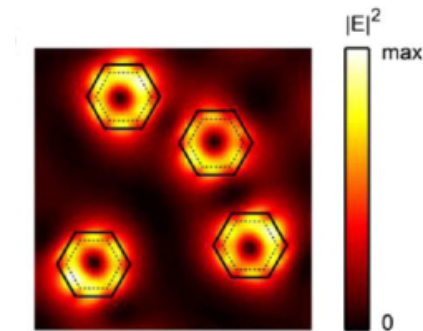
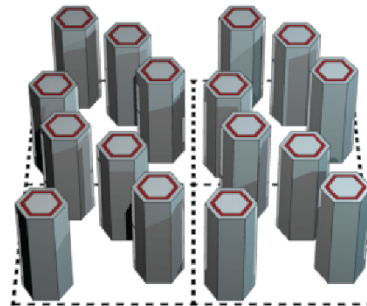
Extraction optimization using Quasi-periodic array



$$\text{F.O.M.} = \frac{\int_{\omega_0 - \Delta}^{\omega_0 + \Delta} \left| \frac{\mathbf{r}}{E} \right|_{Ap}^2 d\omega}{\int_{\omega_0 - \Delta}^{\omega_0 + \Delta} \left| \frac{\mathbf{r}}{E} \right|_P^2 d\omega}$$



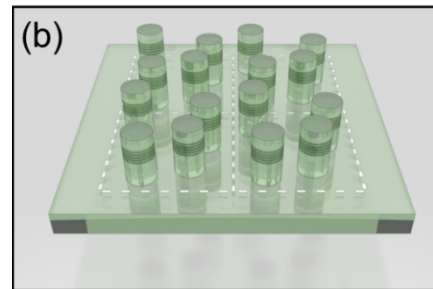
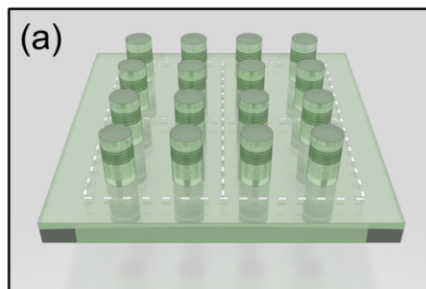
Optimized geometry:



Breaks mirror symmetry

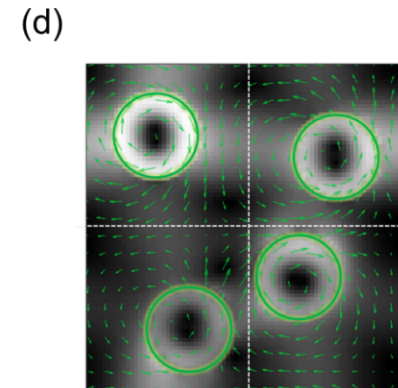
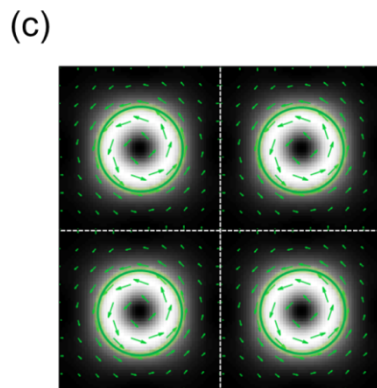
Light extraction control with PC: Quasiaperiodicity

Typical periodic PC



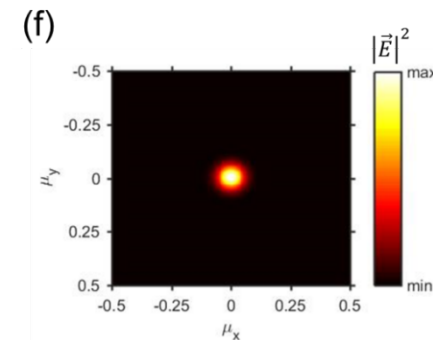
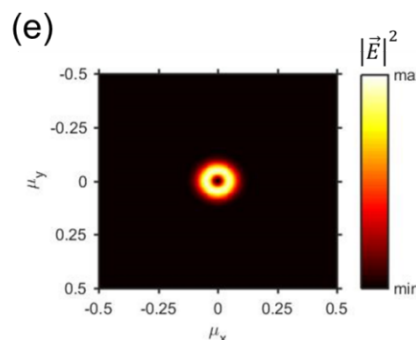
Introduce structural aperiodicity

Symmetric field distribution- doesn't couple to far field



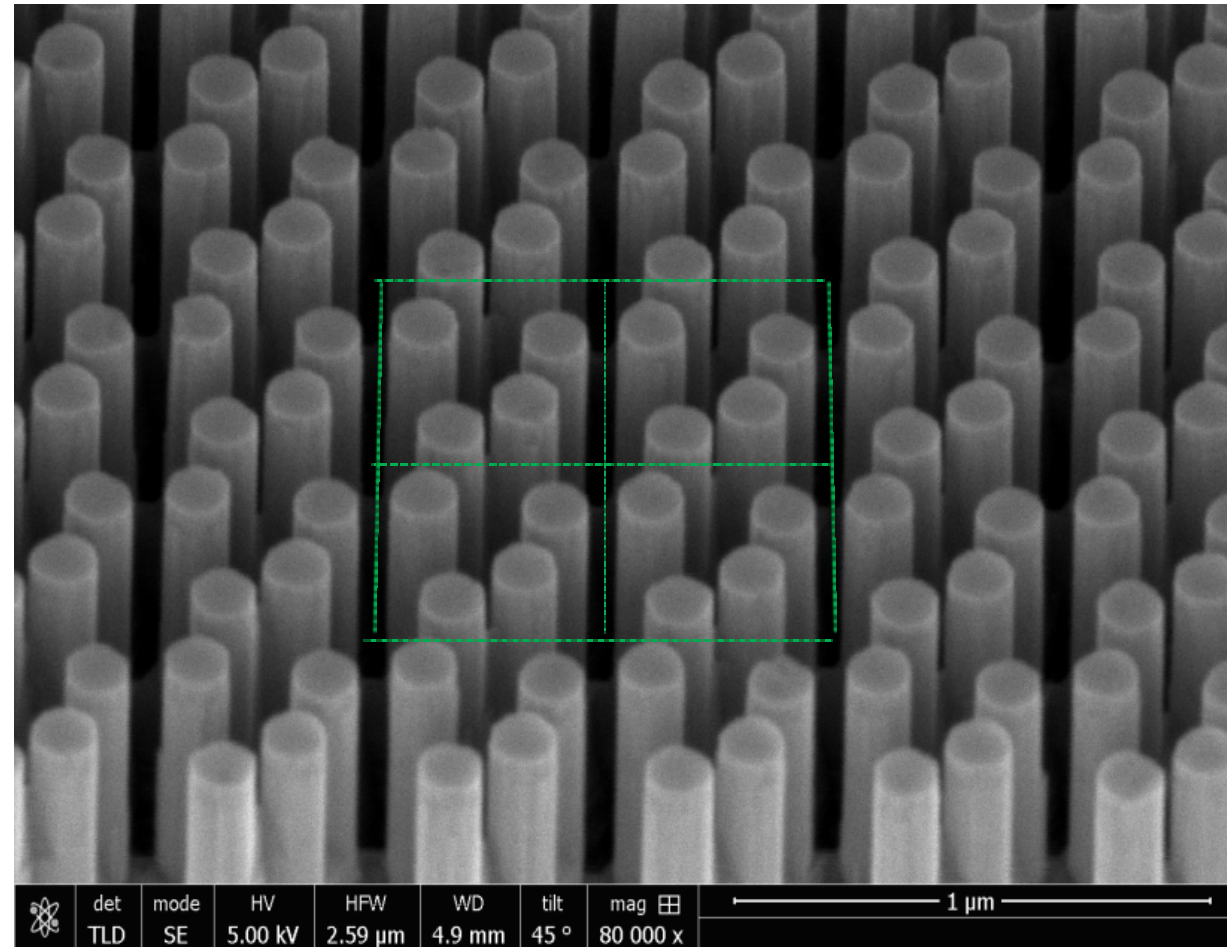
Quasi-aperiodicity induced asymmetric field distribution

Far field distribution has a null in the normal – “donut beam”



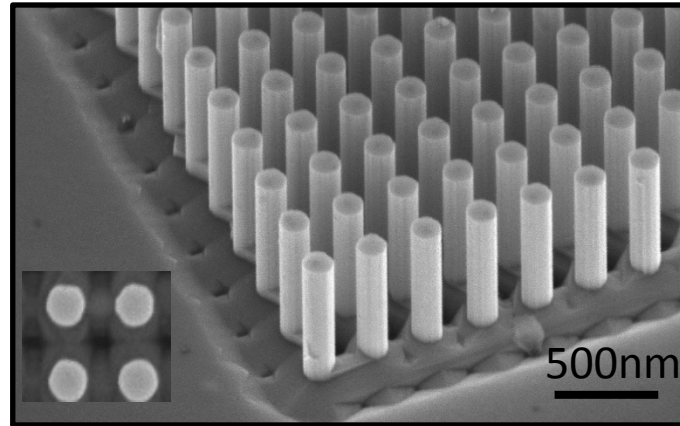
Better far field coupling exhibiting a “Gaussian-like” uniform beam

Quasiaperiodic GaN PhC structures

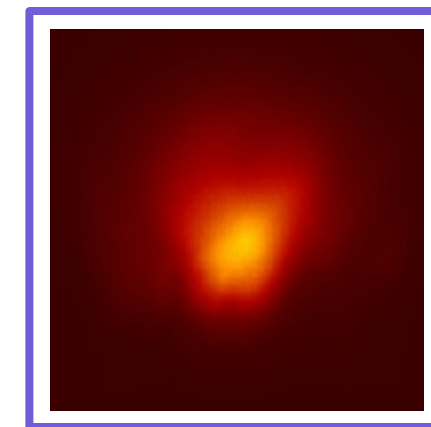
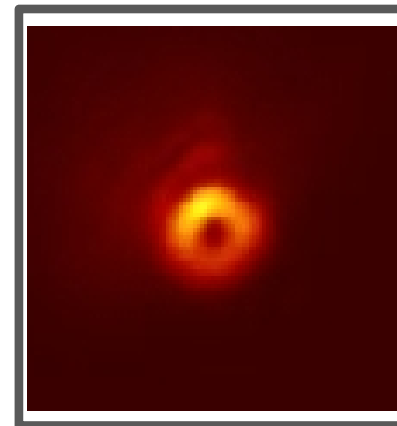
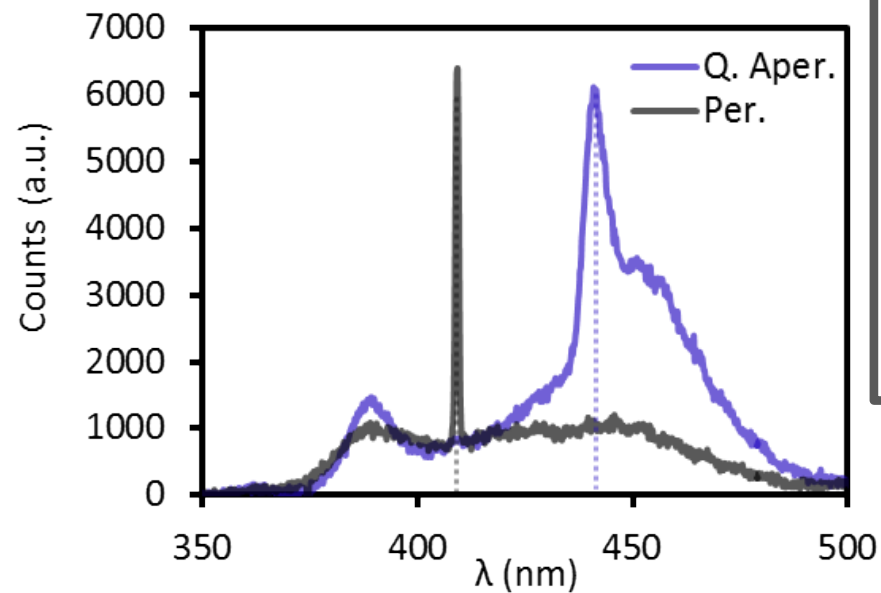
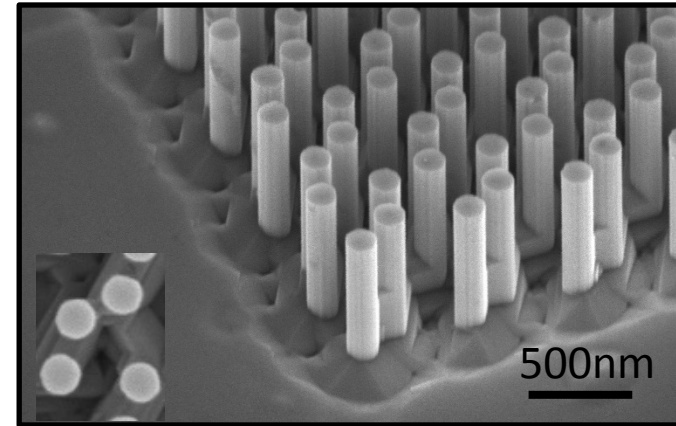


Photoluminescence response

Periodic



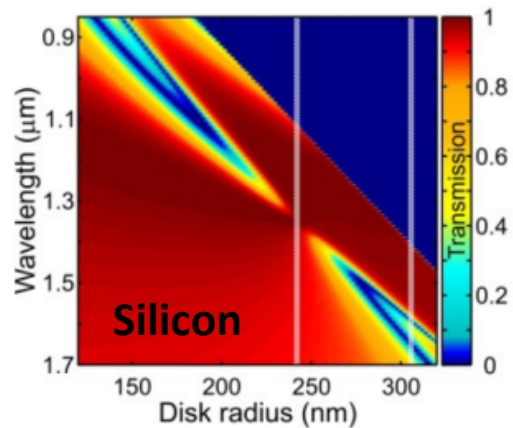
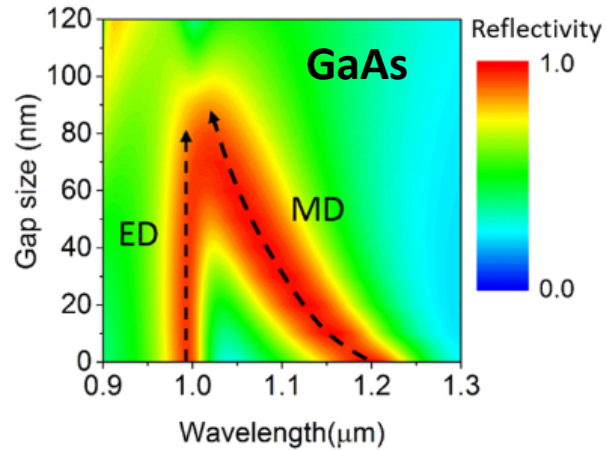
Quasi Periodic



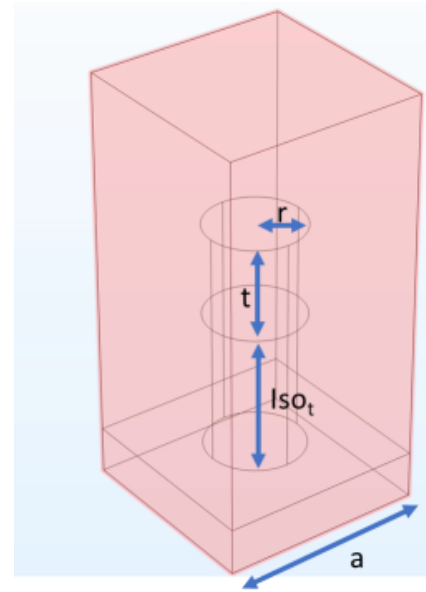
III-Nitrides for Metasurfaces

Huygen's Metasurface Design

Full Transmission

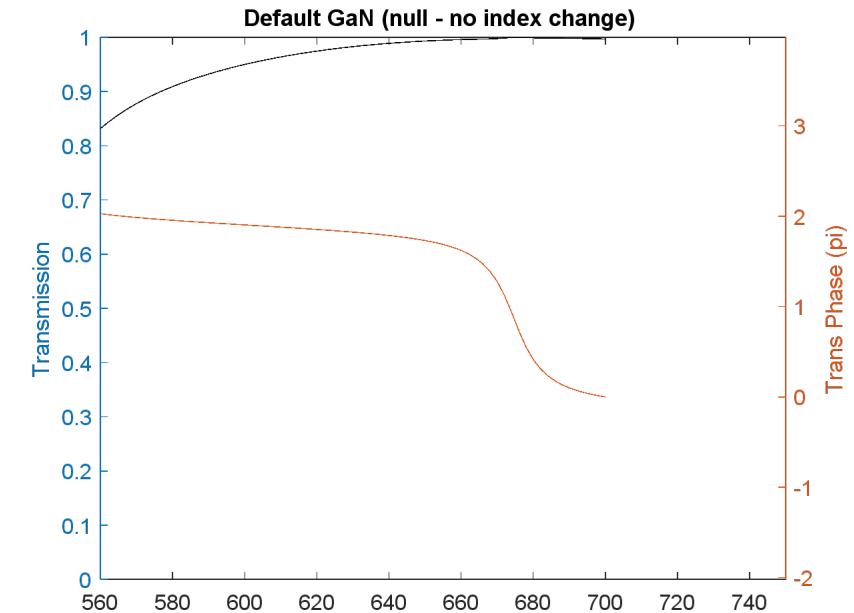


Si Resonator Index = 3.5
Ambient Index = 1.6



$t = 161 \text{ nm}$
 $l_{\text{so}_t} = 300 \text{ nm}$
 $r = 170 \text{ nm}$
 $a = 400 \text{ nm}$

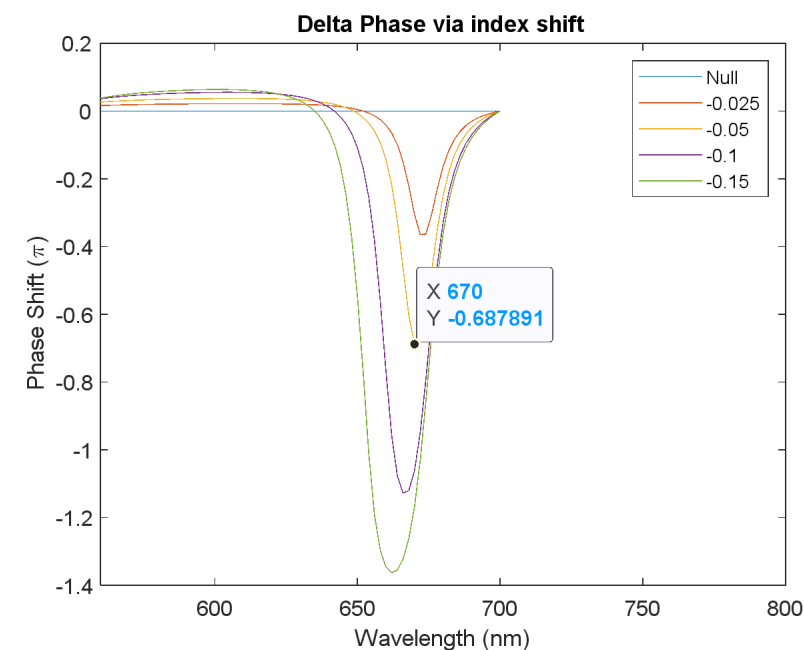
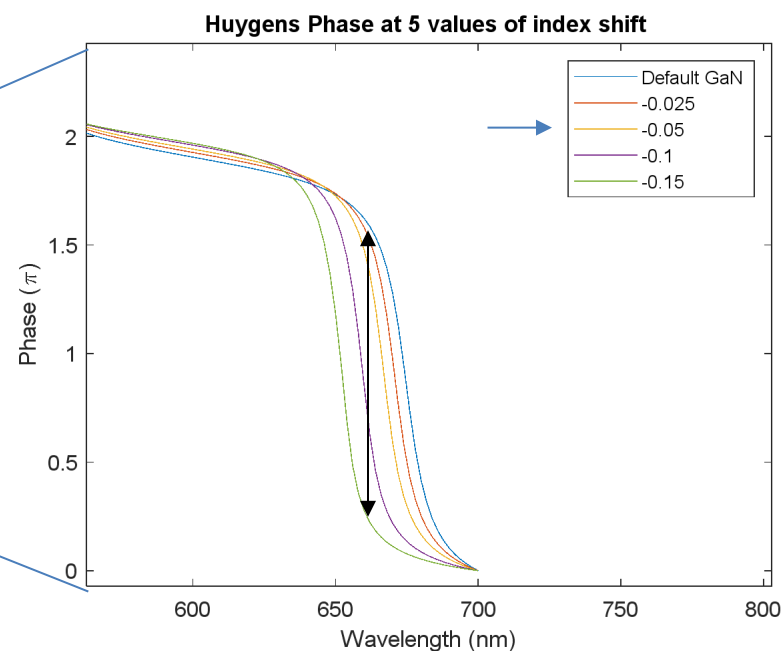
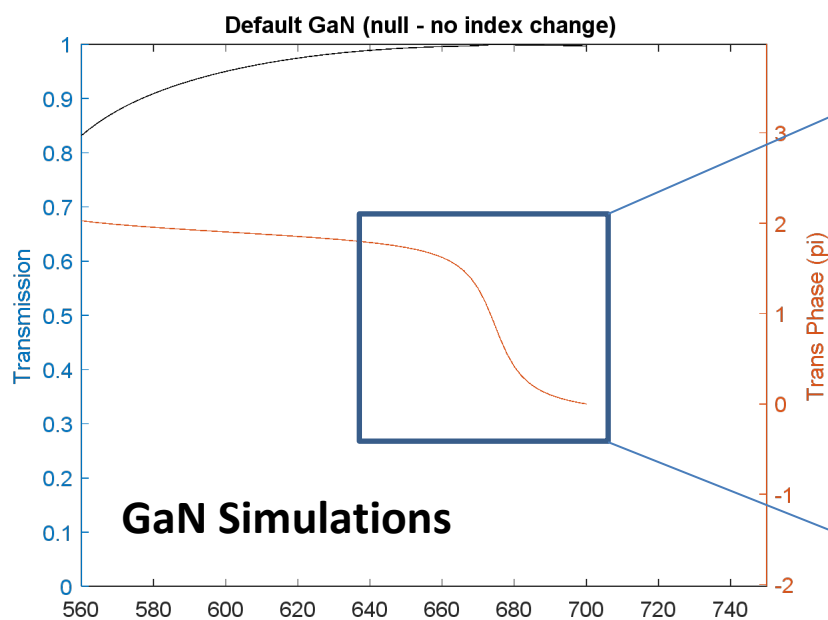
GaN Resonator Index = 2.4
Ambient Index = ?



Target design for isolation layer
index = 1.4 by using porous GaN

Tunable Huygen's Metasurface Design

Tuning the phase response

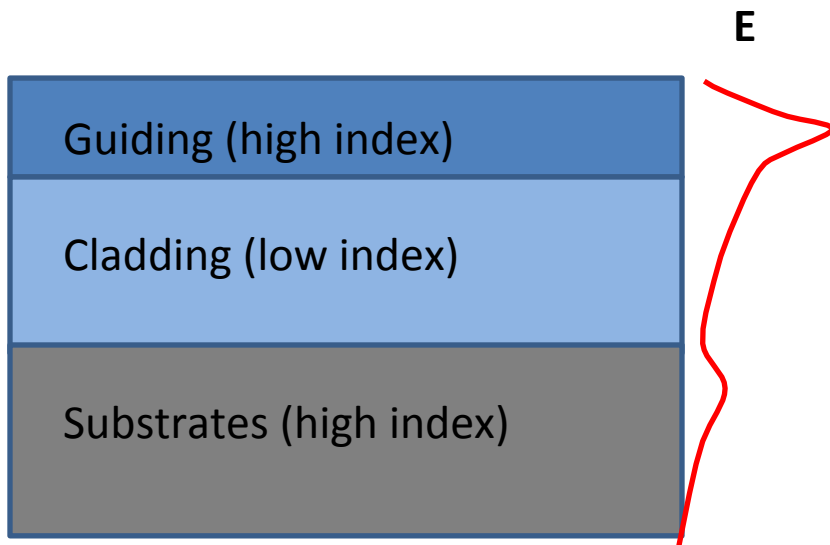


Tuning the resonator index by -0.05 gives

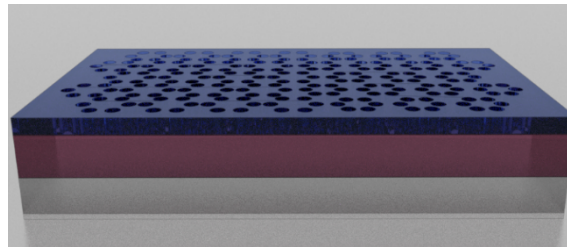
$> \frac{2}{3} \pi$ phase shift at 670 nm

Photonic Devices and Substrate Isolation

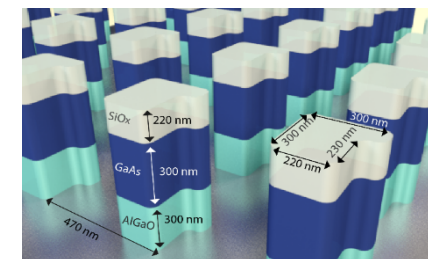
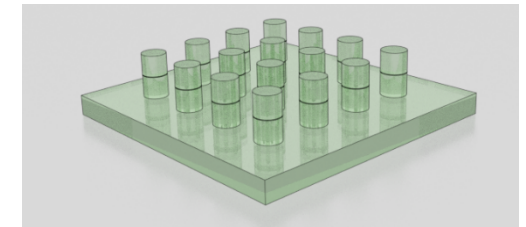
- ❑ Nanophotonic devices need isolation from a high index substrates/template
 - ❑ Electromagnetic field confined to the active or light guiding layer
 - ❑ Low index cladding layer ($n \sim 1 - 1.6$)
 - ❑ Important for photonic crystal or metasurface based devices
 - ❑ Optimal advantage of photonic band structure or scatterer resonance



Hole array PhC



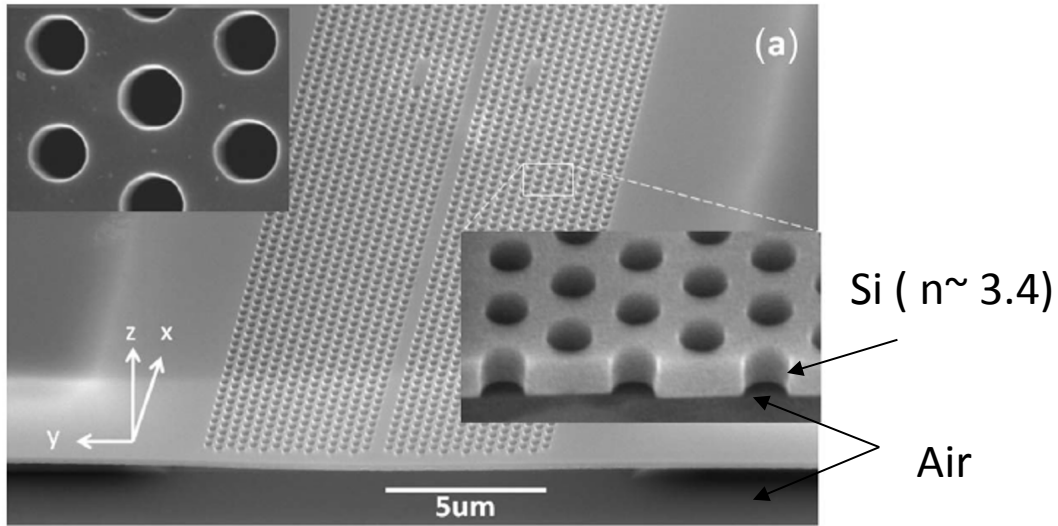
Nanowire PhC



Metasurface

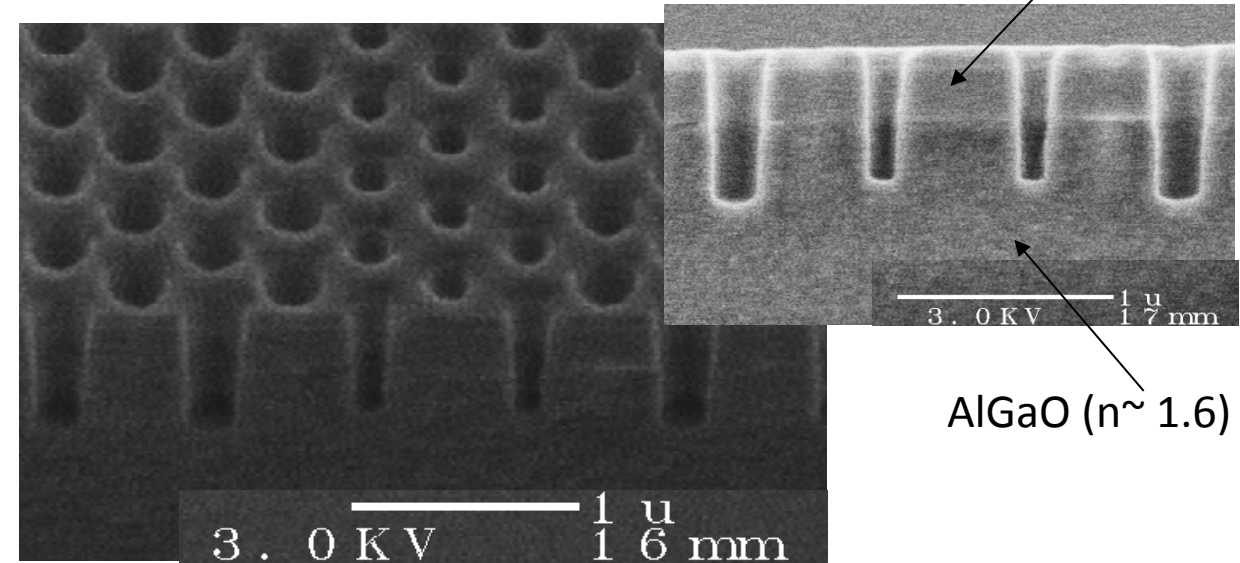
Hole-Array Photonic Crystals

SOI based PhC



T.S.Luk et. al., JOSA. B **28** (6), 1365 (2011).

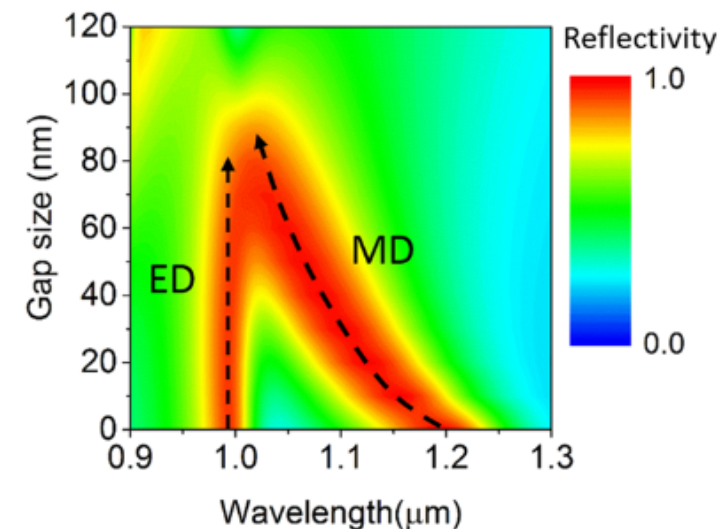
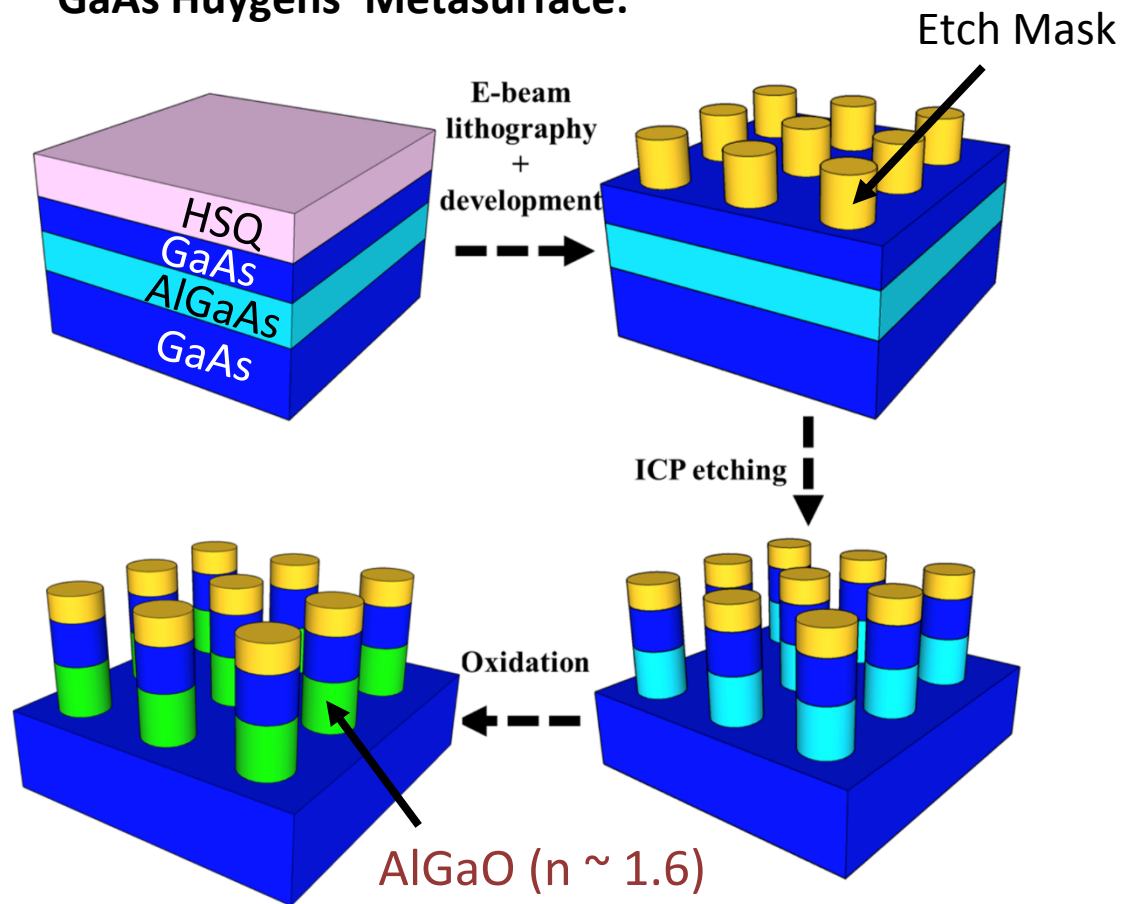
GaAs/AlGaAs based PhC



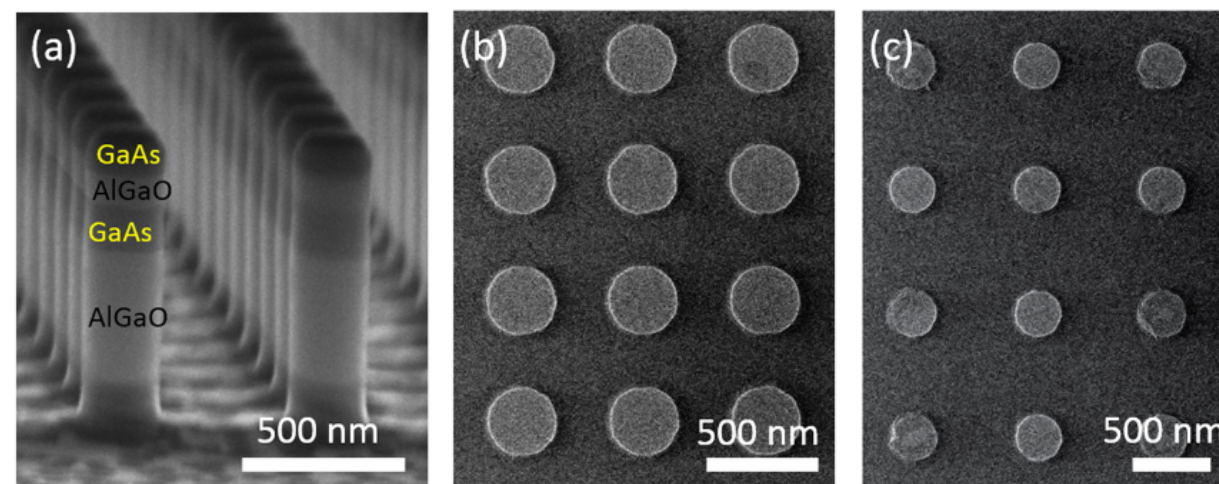
- ❑ Si and GaAs have index ($n \sim 3.4$) can be oxidized to a lower index oxides like SiO_2 or AlGaO with lower index ($n < 1.7$)
 - ❑ Cladding or Etched to form membranes
- ❑ **III-nitrides do not have natural easily oxidizable low index oxides**
 - ❑ Hole array PhC devices are challenging in III-N

Low index layer in III-V for Optical Isolation

GaAs Huygens' Metasurface:



Design for overlap of ED and MD (Huygens' condition)

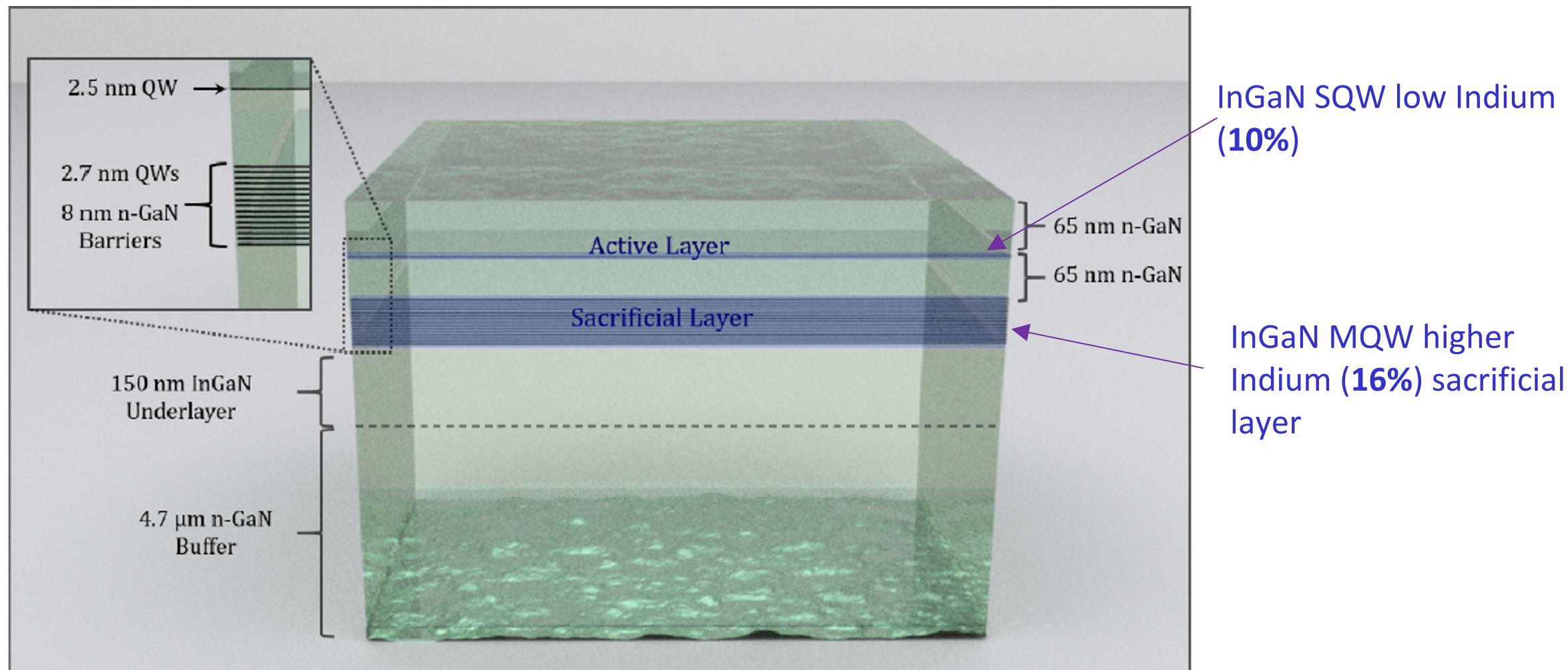


III-nitrides do not have natural easily oxidizable low index oxides – Substrate Isolation challenging in III-N

Figures adapted from: Liu, S. Nano Lett. 16.9: 5426, (2016)

Liu, S. Nano Lett. 2017, 17, 4297–4303

III-Nitride Membrane PhC : One approach

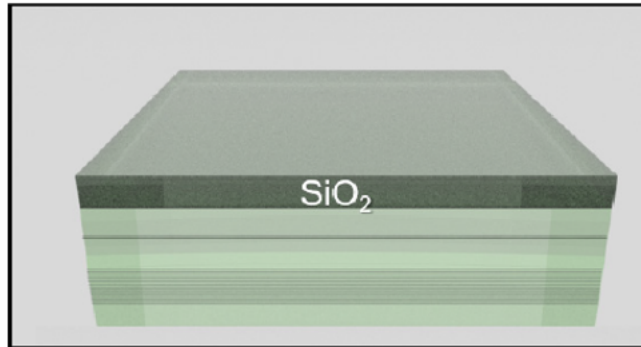


Epi Structure with a bottom sacrificial MQW layer

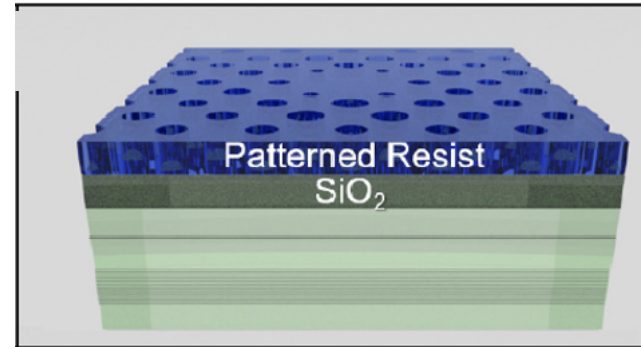
Figures reprinted/adapted from Anderson, P. D.; Fischer, A. J.; Koleske, D. D.; Gunning, B. P.; Subramania, G., *Optical Materials Express* **2018**, *8* pp3543-3550.; Copyright 2018 OSA.

Fabrication steps

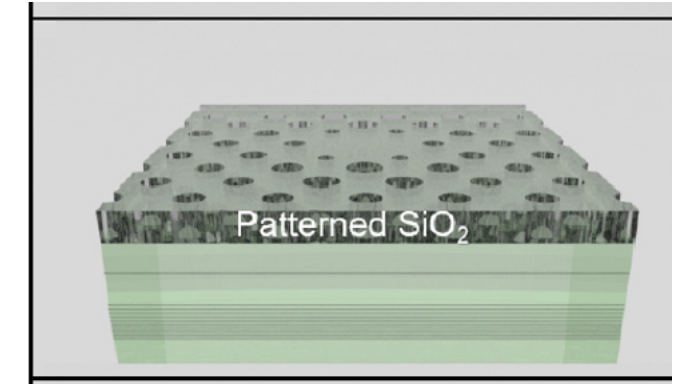
Deposit SiO₂ hardmask



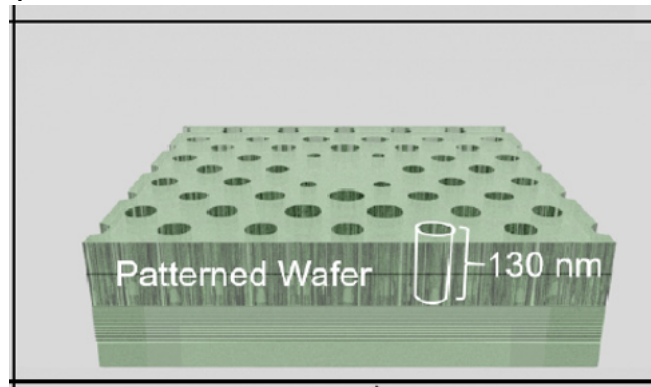
EBL pattern in PMMA



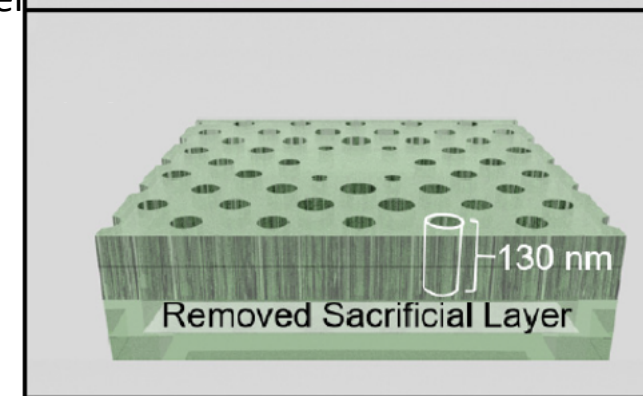
Etch Hardmask



Etch III-nitride active layer through patterned hardmask

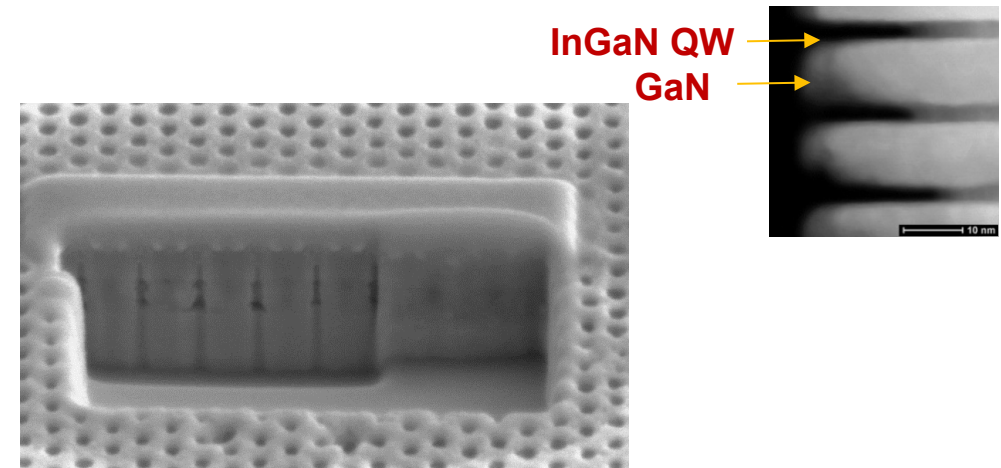
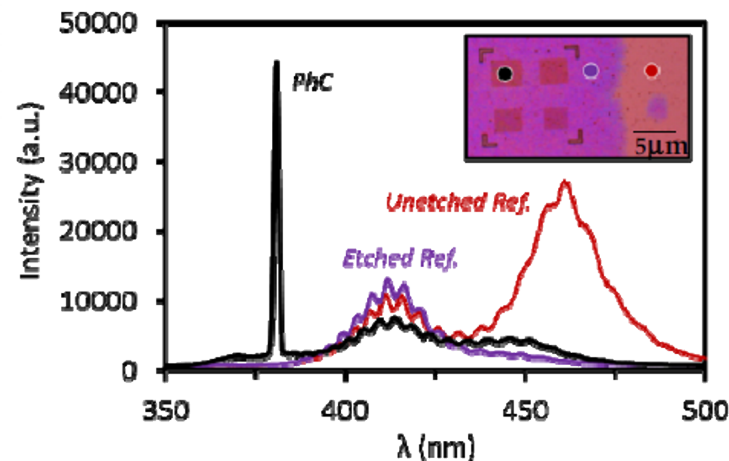
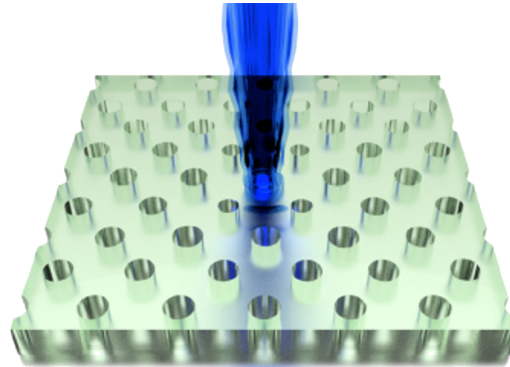
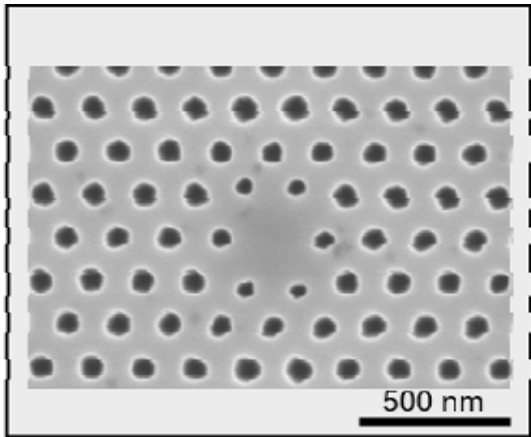
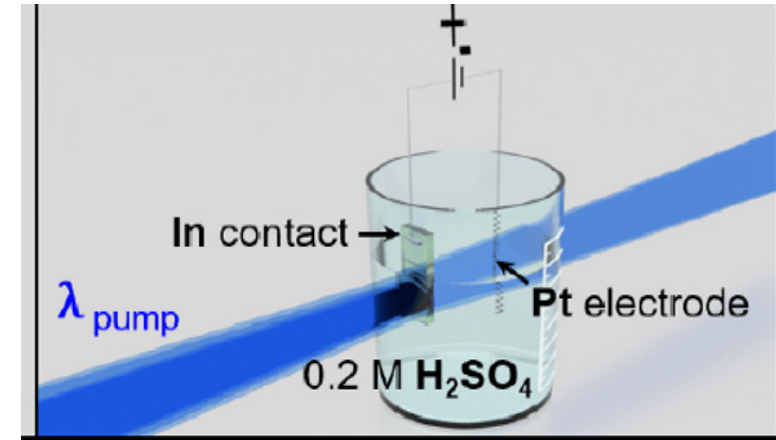


Remove sacrificial multiquantum well layer



Photoelectrochemical Etching (PEC)

- Electrochemical etch is performed under laser illumination
- Laser wavelength is selected such that it is shorter than the energy gap of 16% InGaN but longer than that of 10% InGaN
- Selective etching of 16% InGaN MQW
- Creates low index cladding layer

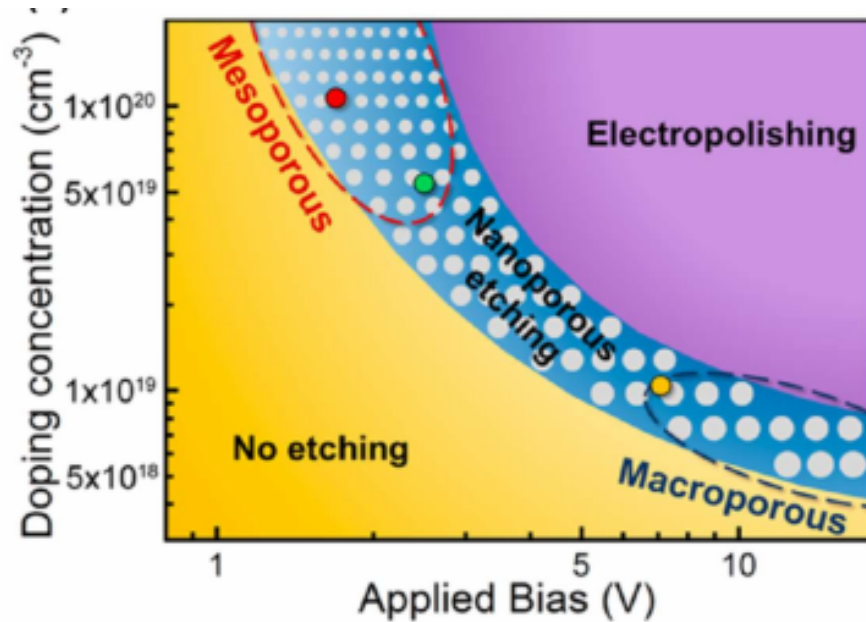


- ❖ Removal of InGaN QWs by PEC reduces cladding layer index but procedure is challenging
- ❖ Hard to scale for thicker cladding

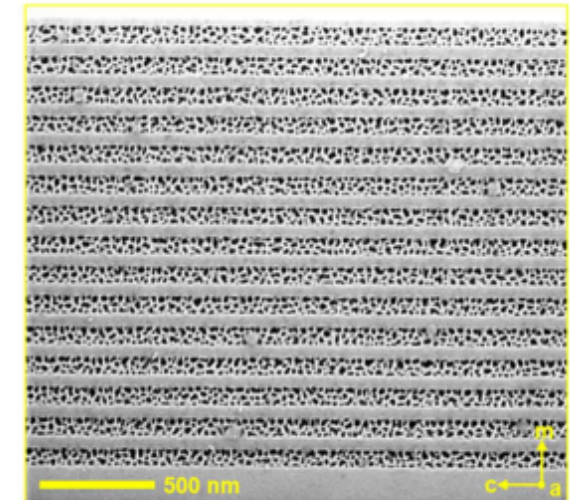
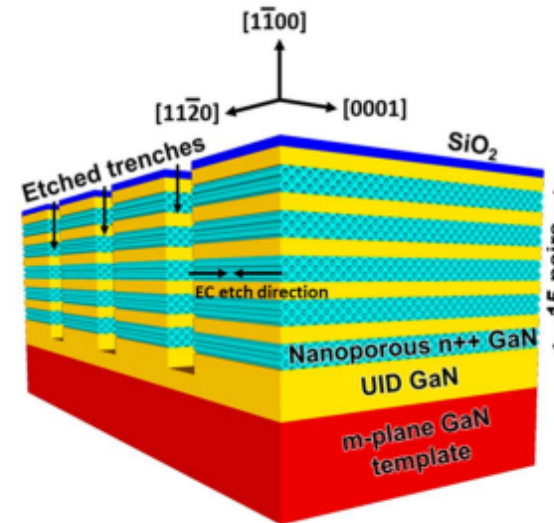
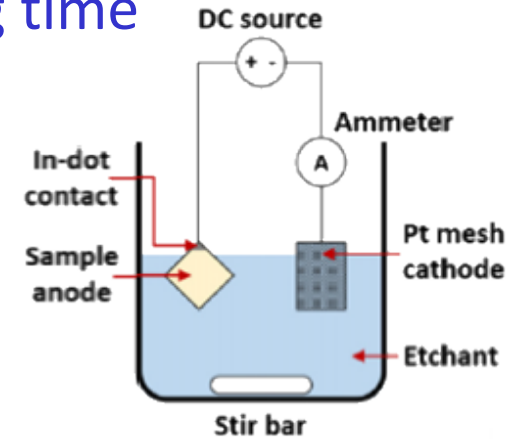
Nanoporous III-N Low Index Layer

Electrochemical (EC) etching of doped III-N layer in chemical bath

- Etching conditions - applied voltage, doping concentration, etching time



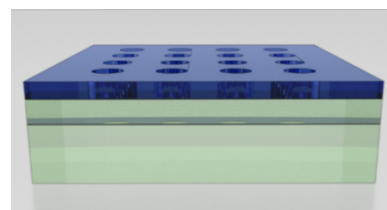
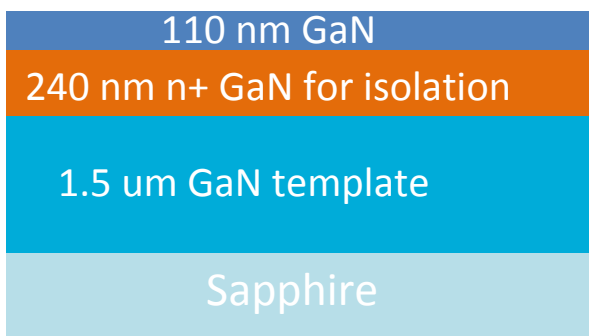
Feezell Group (UNM)



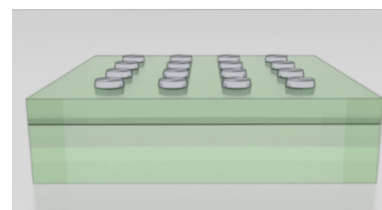
Approach is flexible and scalable

- Enables control of porosity a through control of etch conditions

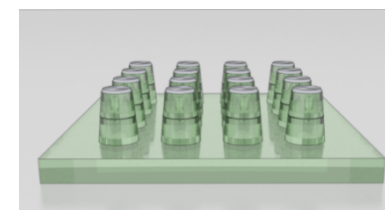
III-N Nanowire Array Resonators



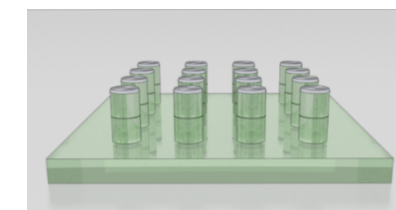
EBL pattern
in PMMA



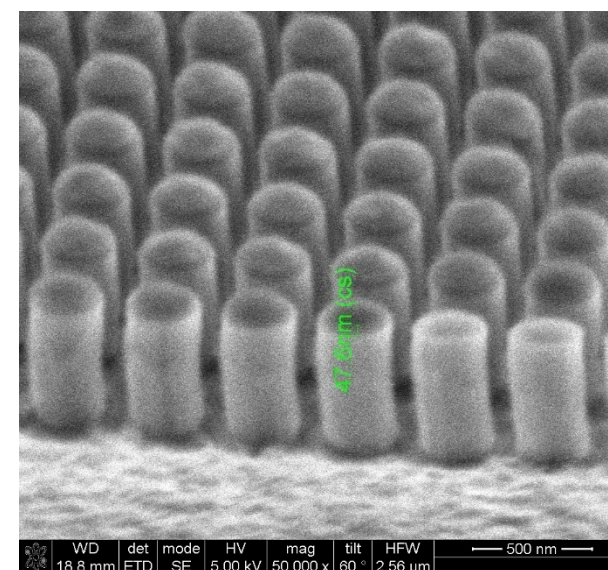
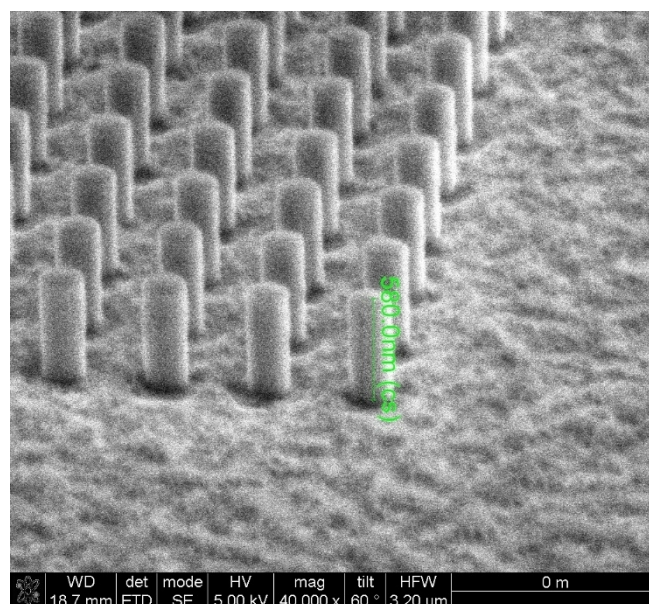
Ni evaporation
and lift-off



Cl_2 based dry etch



KOH based wet etch

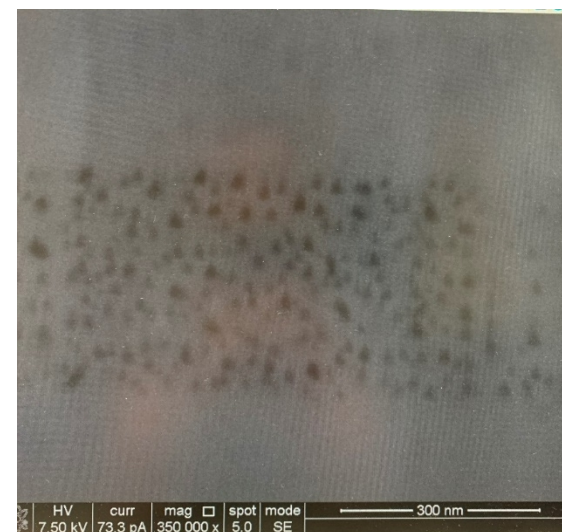
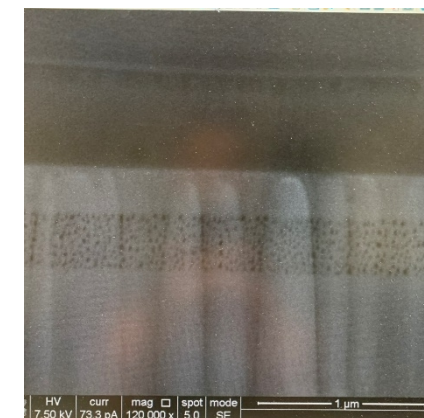
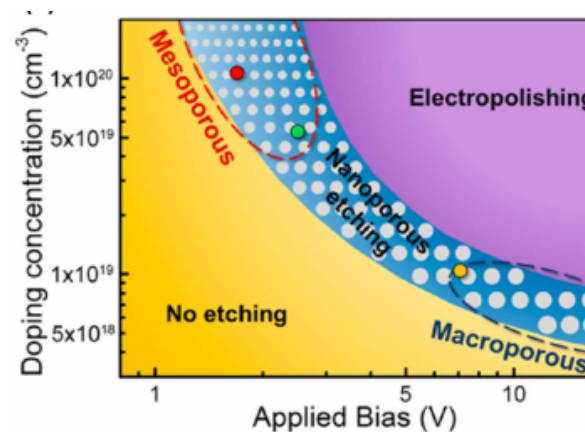
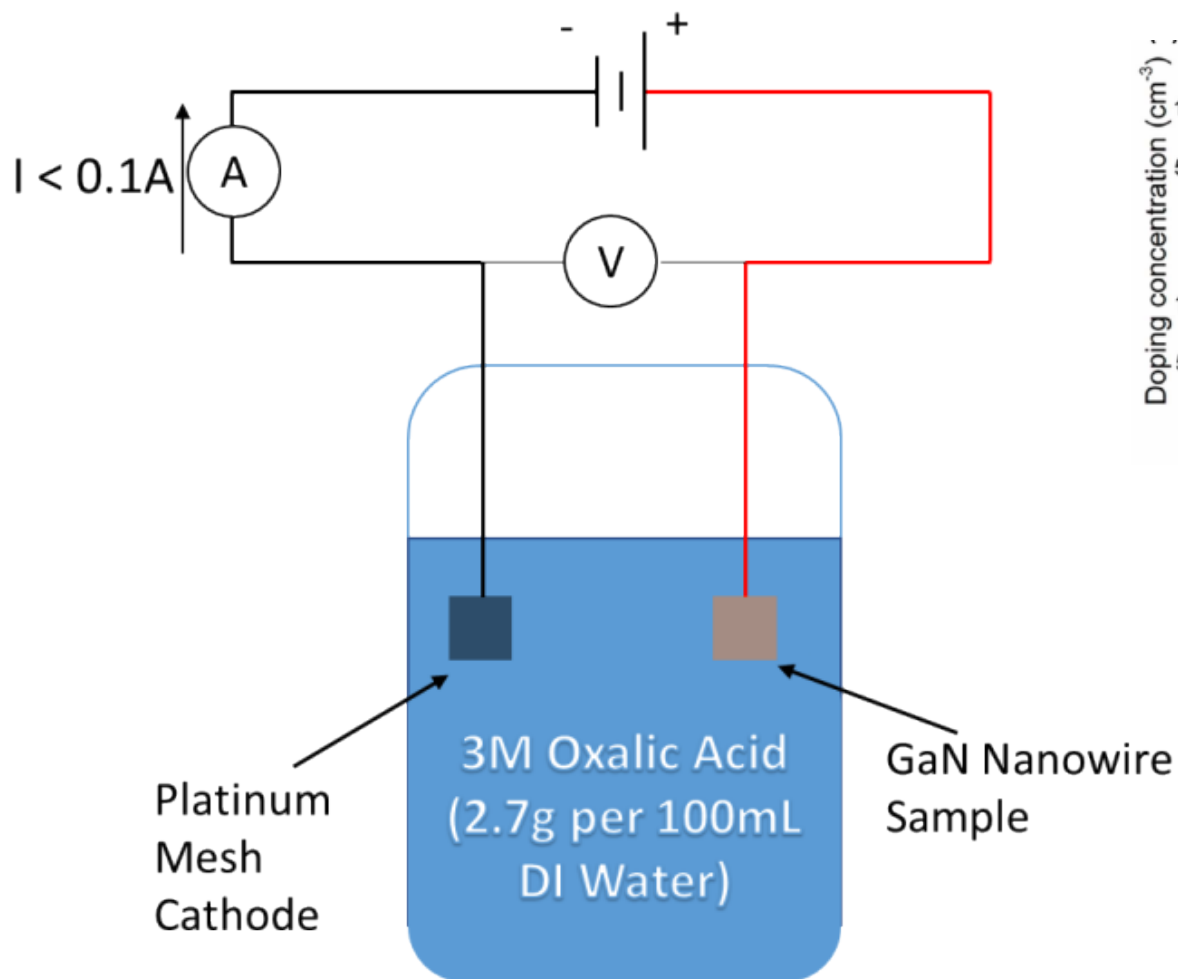


Demonstration of cylindrical GaN nanowire array with smooth side walls

Electrochemical Etching Process

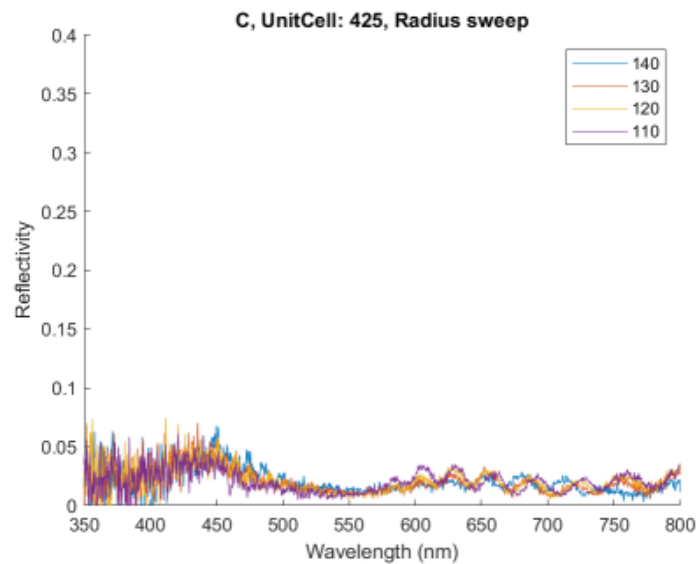
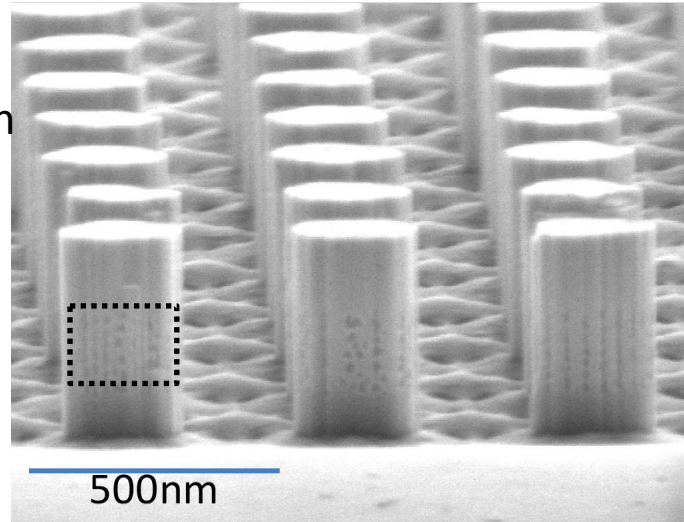
Electrochemical etching of doped GaN layer in acidic solution (Z. Meinelt / D. Feezell (UNM))

- Results in nanoporous region with lower effective refractive index

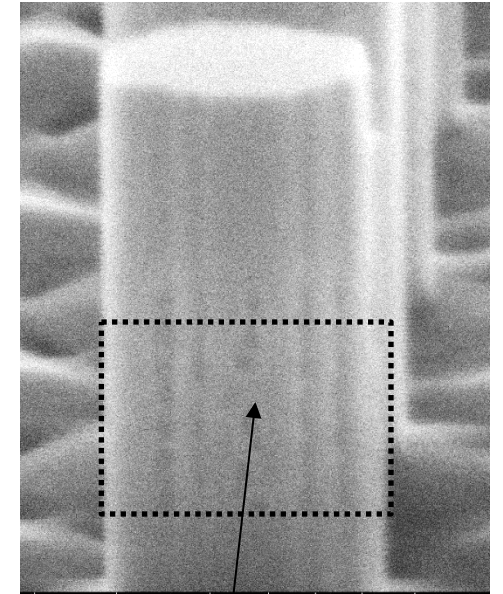
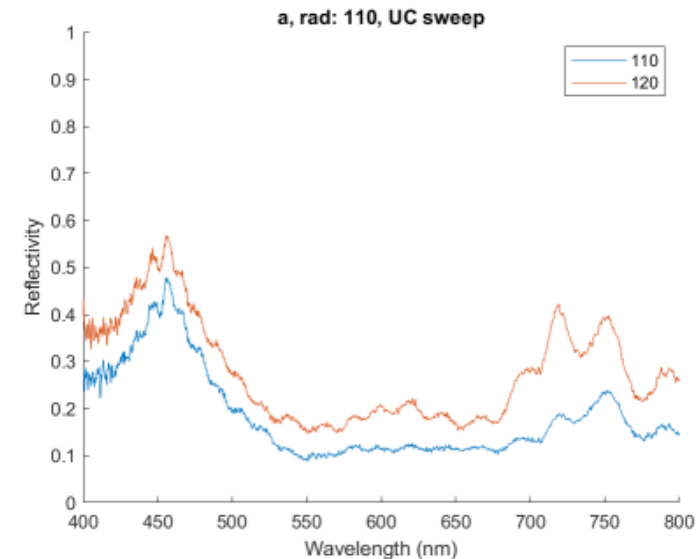
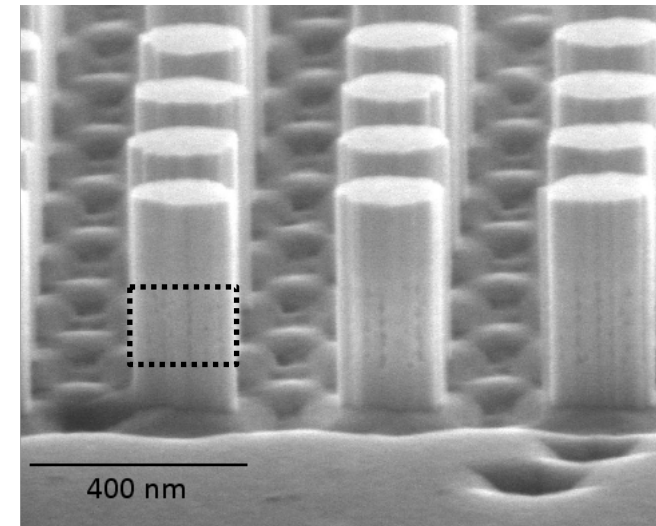


Electrochemical Etch of Nanowire Resonators

Etch voltage: 6V



Etch voltage: 7V



Nanoporous GaN region
($n \sim 1.2 \times 10^{-19} \text{ cm}^{-3}$)

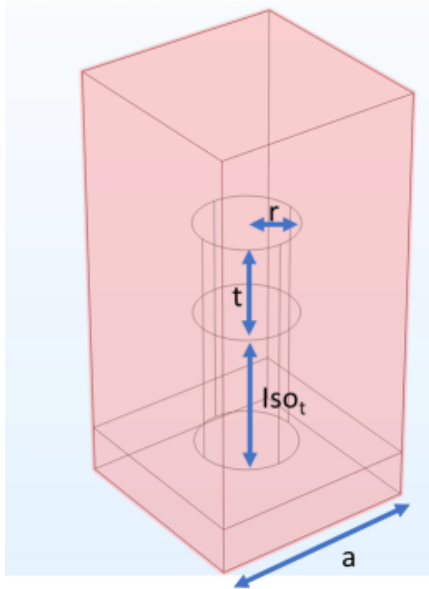
EC etching shows reduction in refractive index evidenced in optical response

Porosity Characterization

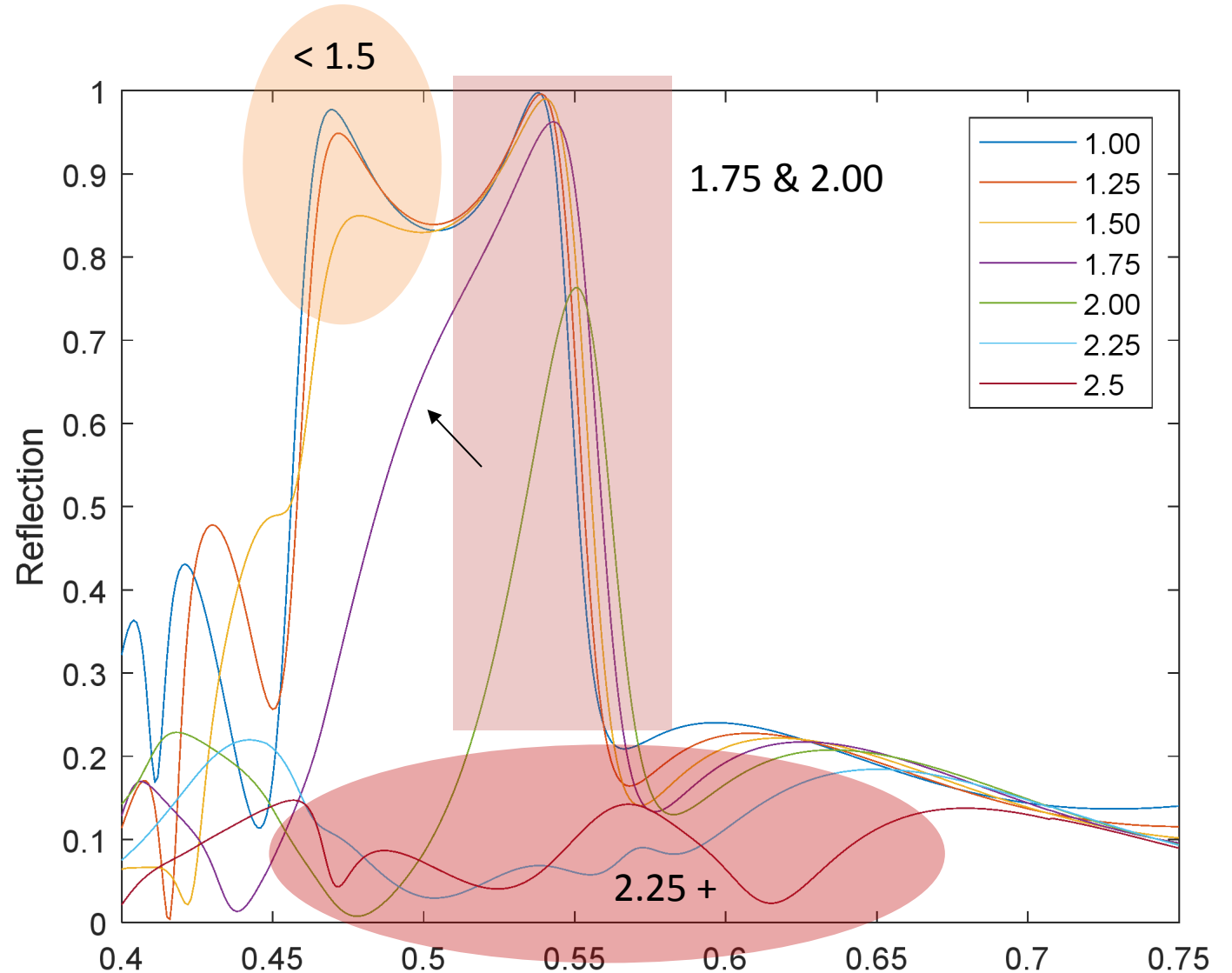
Targeting an effective index of < 1.5

Huygens' MS has complicated optical response when isolation layer index is incorrect

MS with separated ED and MD is used to guide fabrication



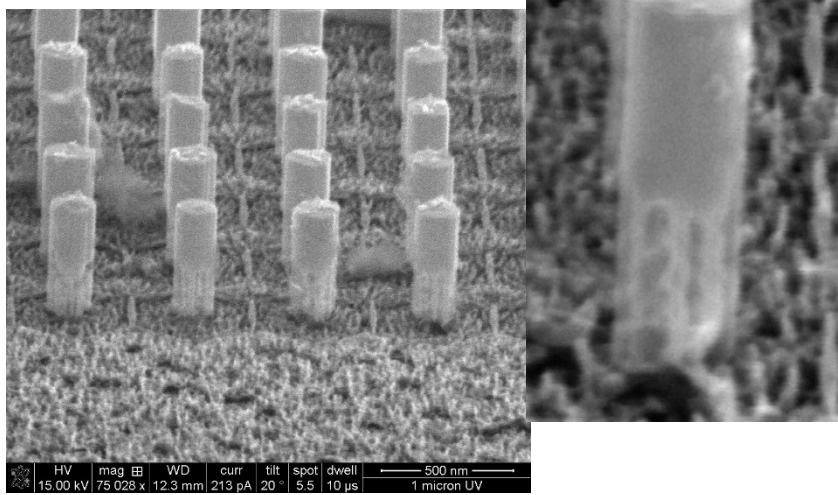
$t = 200 \text{ nm}$
 $\text{Iso}_t = 300 \text{ nm}$
 $r = 105.2$
 $a = 440 \text{ nm}$



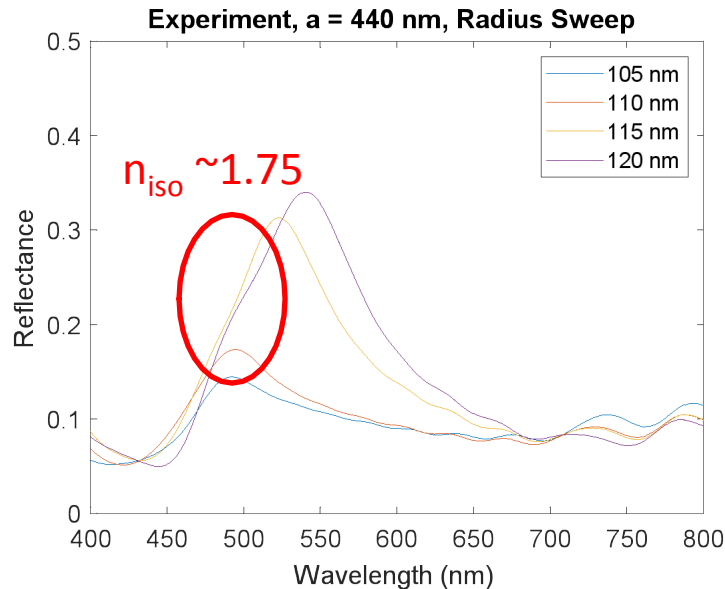
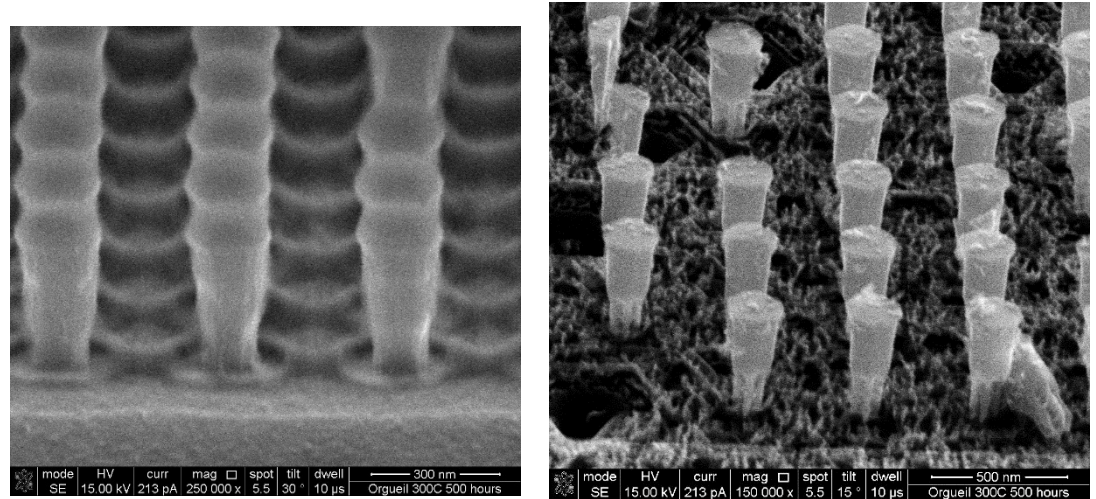
Porosity Characterization

(24 V, 12 min)

EC etch

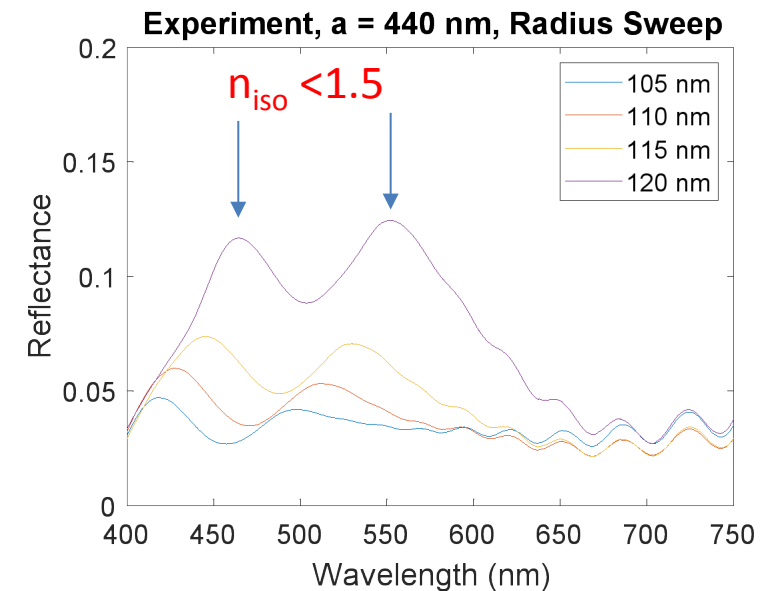


Blanket wet etch (KOH) + EC etch



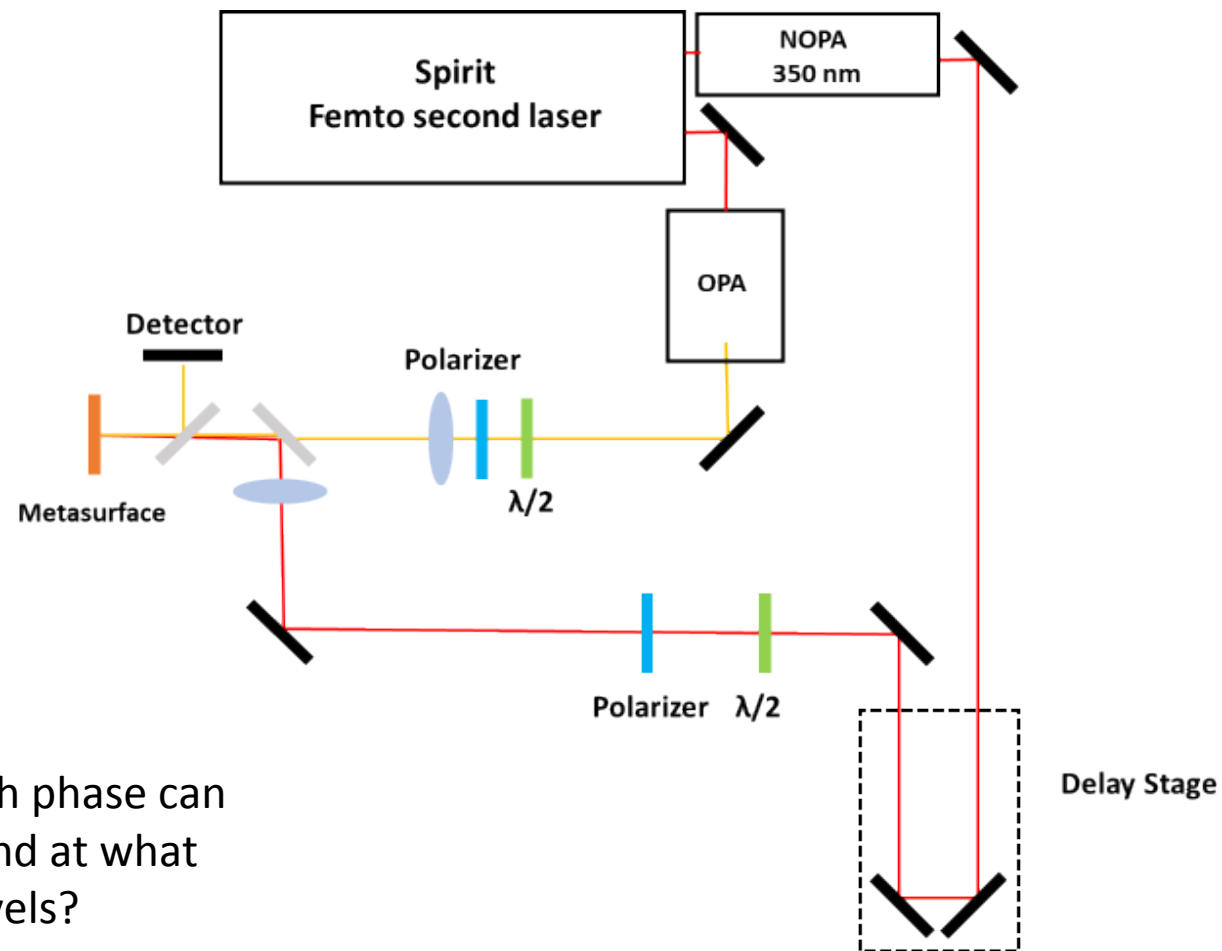
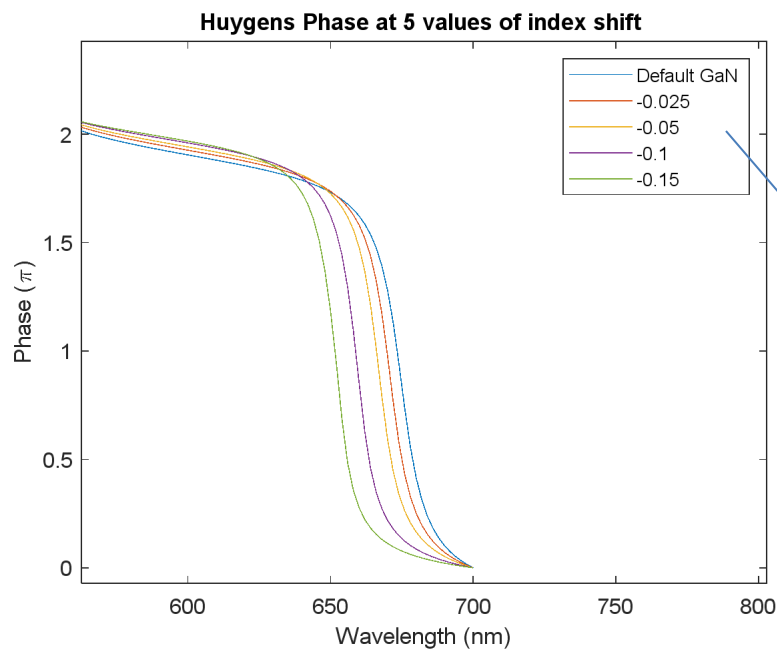
Using our characterization design, the nanowire resonators show a clear reflectivity response enabling porosity/effective index characterization.

With the additional wet etch the porosity target enabling our Huygens' MS has been achieved!



Tuning Characterization

Characterize the transmission and phase response of the Huygens' MS as the resonator index is tuned (via ultrafast optical pumping)



How much phase can we get, and at what power levels?

Summary

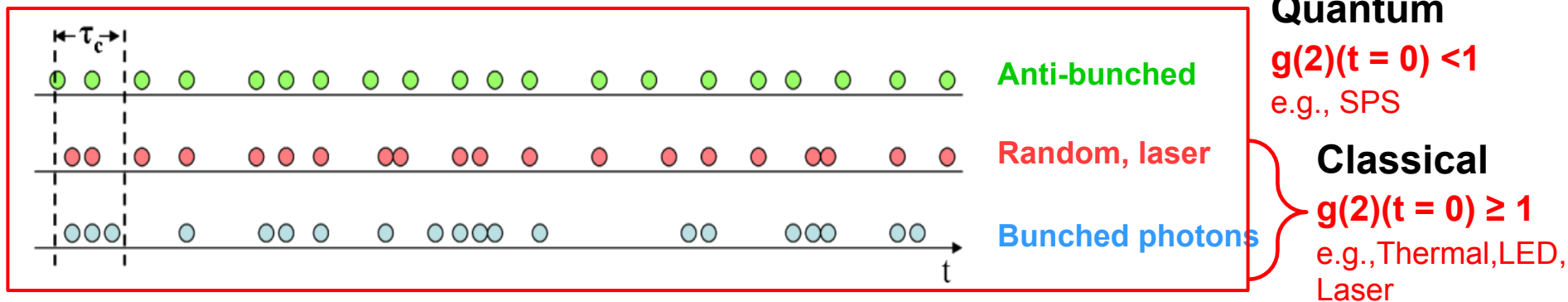
- **Role of nanostructuring in light control**
- **2D Nanowire Photonic Crystals exhibits vertical lasing emission**
 - Can modify emission profile
- **III-Nitrides for Metasurfaces can enable tunable response**
 - Substrate isolation cladding will be important; EC etch provides an elegant route
- **Quantum Dots for Single Photon Sources (time permitting)**
 - Photo electrochemical etch can enable size controlled QDs

Thank you for your attention!

Back up slides

Quantum Dots for Single Photon Sources

Single Photon Source: Photon Statistics



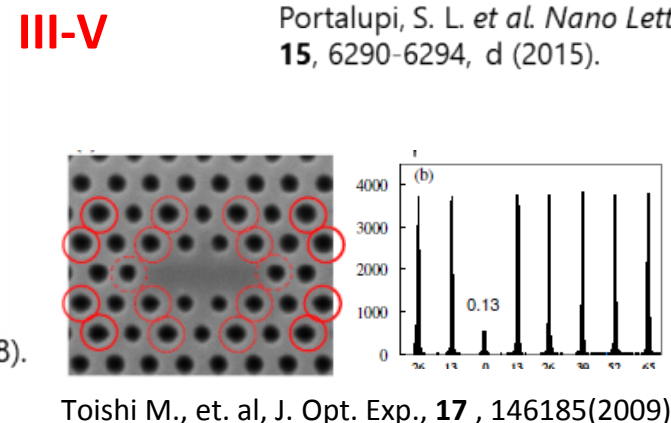
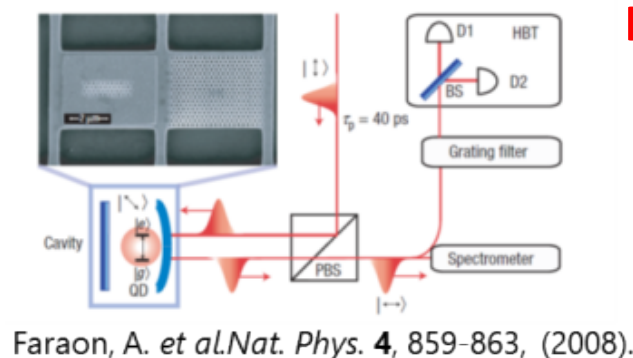
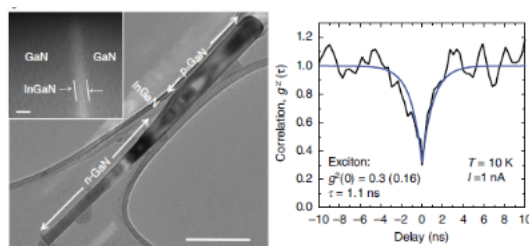
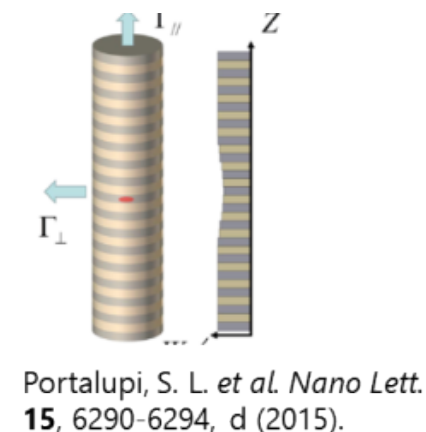
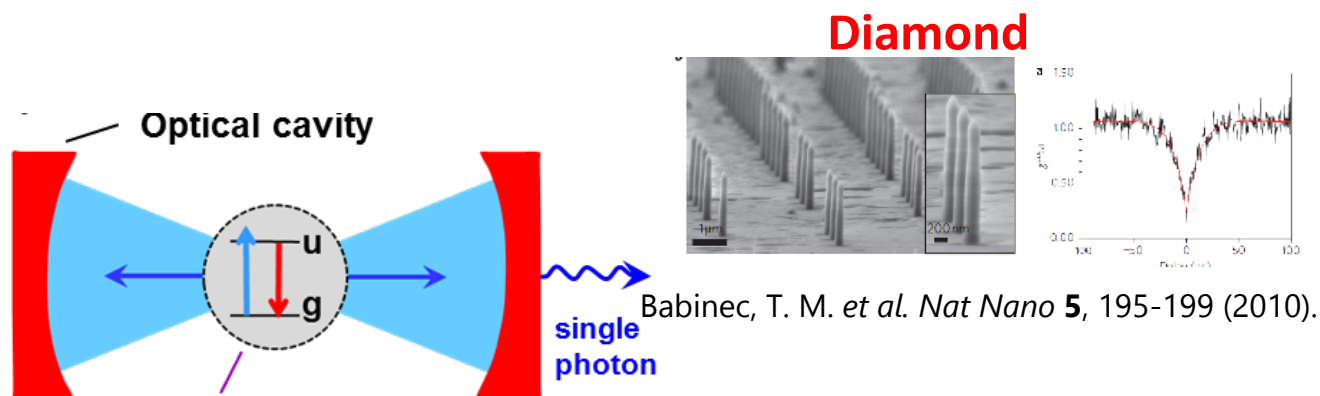
source: J.S. Lundeen

- **Quantum communication**
 - High bit rates than attenuated laser
 - Long distance secure comm.
 - Quantum repeaters (with indistinguishability)
- **Quantum metrology**
 - Measurement of low absorption
 - Single molecule level detection (in combination with high efficiency SPD)
- **Quantum computing**
 - Qubit operations
 - 'flying' qubits

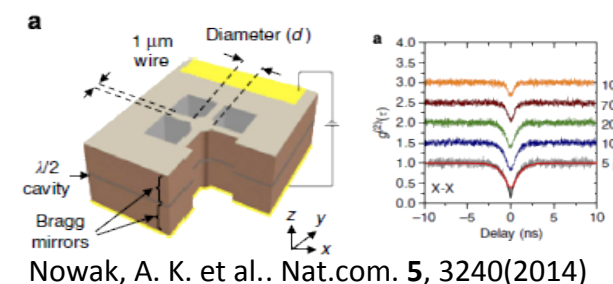
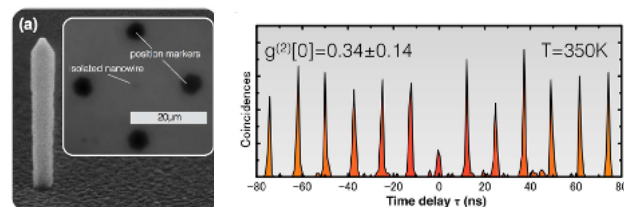
Desired properties of practical single photon sources (SPS)

- ❖ Near 100% probability of emitting single photon : $g_2(0) \sim 0$
- ❖ High single photon rates
- ❖ High extraction or collection
- ❖ Mode quality
- ❖ Room temperature operation
- ❖ Electrical injection
- ❖ ~ 100% indistinguishability

Single Photon Source based on Quantum Dots



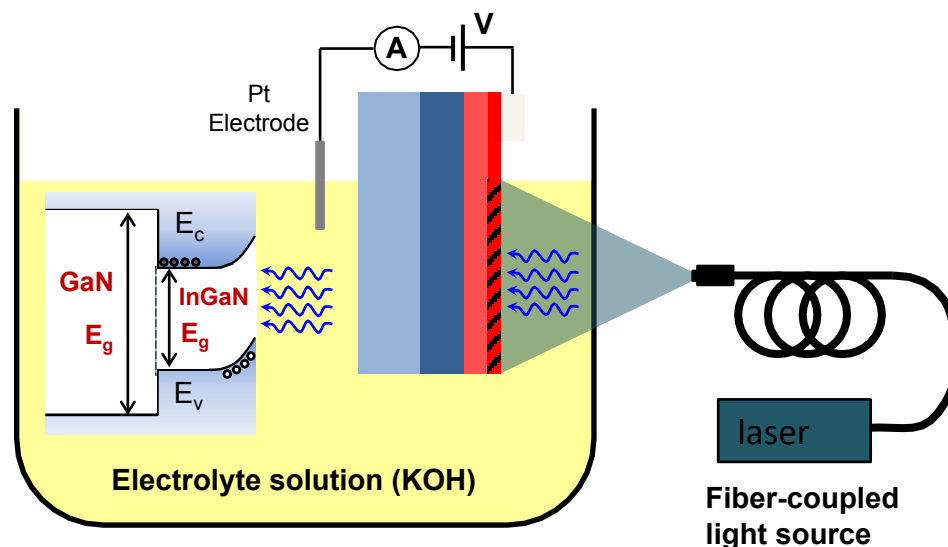
III-N



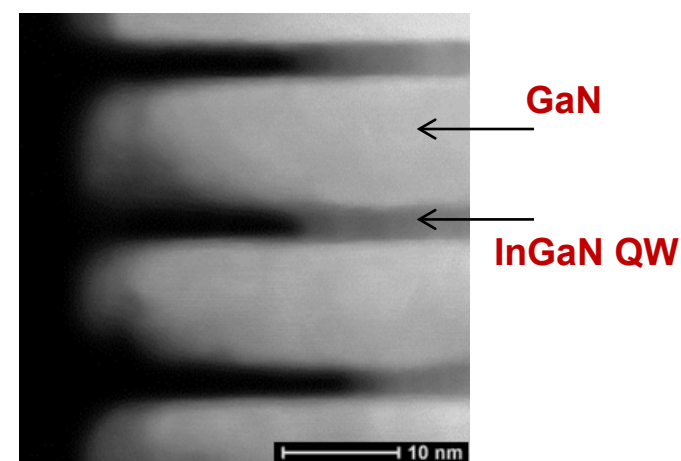
Motivation for InGaN Quantum Dot SPS

Room temperature, electrically-injected, chip-scale single photon source

- Large exciton binding energy so can enable room temp. operation
- Path to electrical-injection/chip-scale integration
- **Photoelectrochemical (PEC) Etching:**
 - Very few wet etches work for III-nitrides
 - **Band gap selective (Etch InGaN over GaN)**
 - Dopant selective, light intensity dependent, etch current can be monitored
 - Laser or lamp excitation (Xe arc lamp, tunable ps Ti:S)
 - KOH (~0.1M) typically used as electrolyte



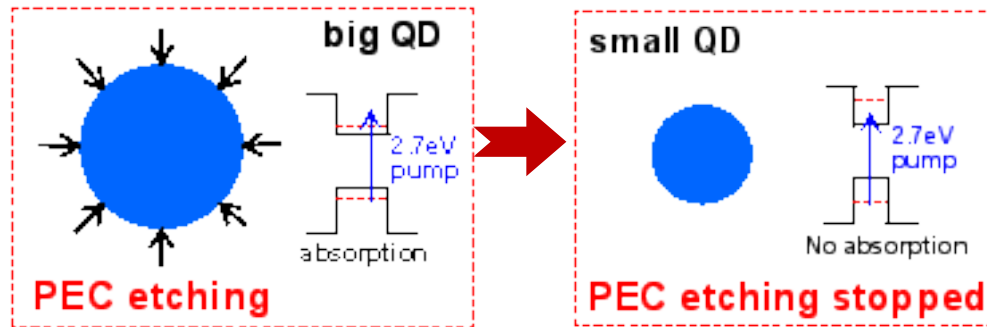
PEC etched InGaN/GaN QWS



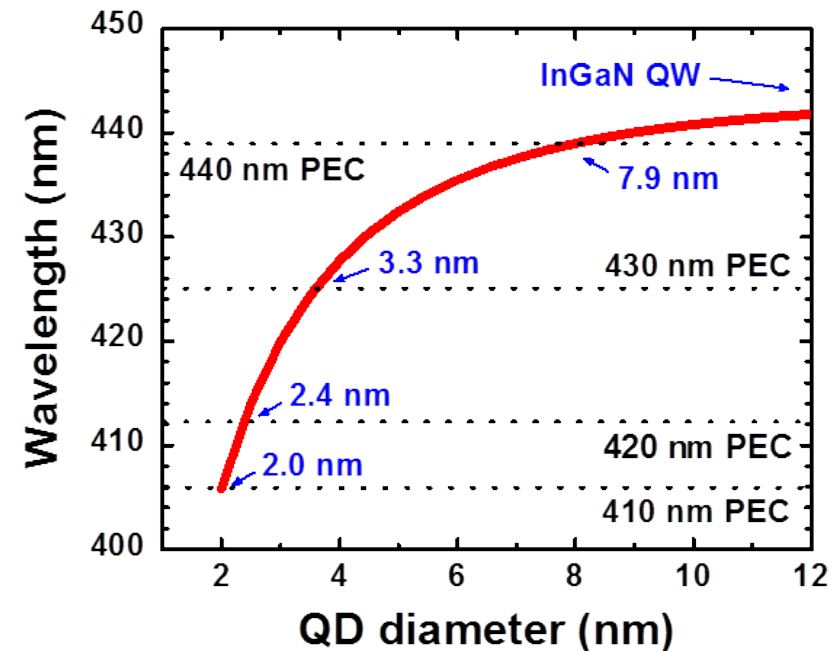
Fabrication of InGaN QDs via PEC etching

Quantum Size Control: Use size quantization to control QD size

Self-limiting PEC etch process:



- For QDs, band gap depends on size
- As etch proceeds,
 - QD size gets smaller, band gap goes up
 - Etch terminated for $E_g > E_{\text{photon pump}}$
- **Self-terminating etch process**
- **Band gap selective**
- **QD size depends on PEC wavelength**
- **Monodisperse QD distributions ??**

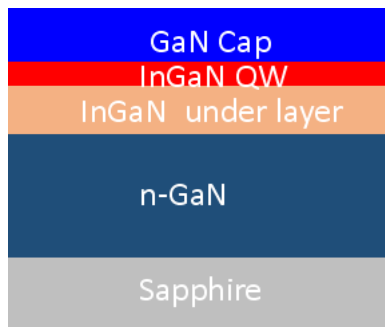


G. Pellegrini, et al., Journal of Applied Physics
97, 073706 (2005).

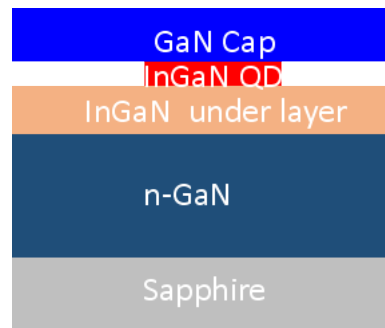
Xiao, X. *et al.* Quantum-Size-Controlled Photoelectrochemical Fabrication of Epitaxial InGaN Quantum Dots. *Nano Lett.* **14**, 5616-5620, doi:10.1021/nl502151k (2014).

Transmission Electron Microscope Images

Capped InGaN QW

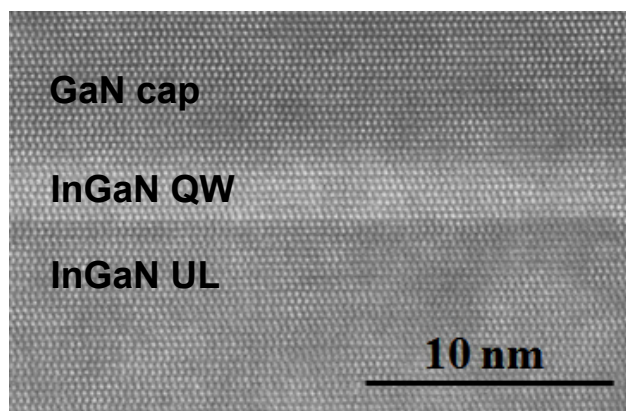


InGaN QD

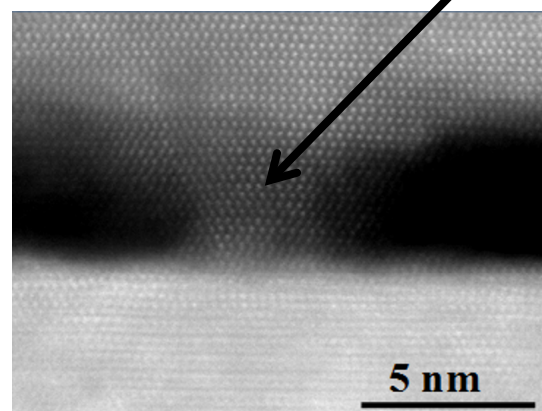


- High-angle annular dark-field (HAADF) TEM images
- Sample etched at 420 nm
- EDX mapping shows that dots are InGaN
- InGaN QDs are epitaxial to the underlying GaN
- 2% InGaN underlayer + GaN cap
- GaN cap provides partial passivation

before PEC etch

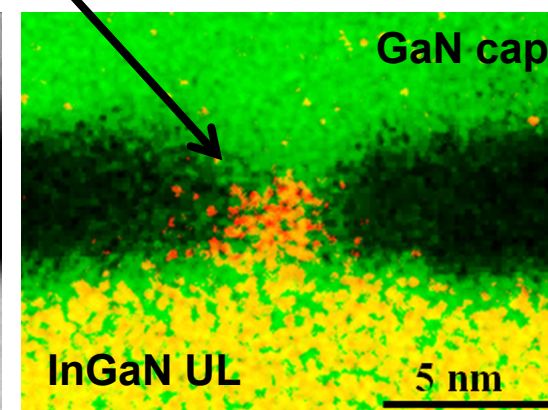


after PEC etch



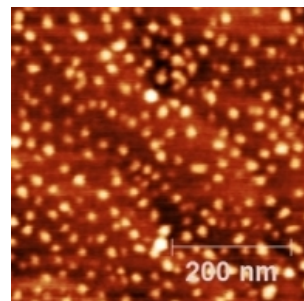
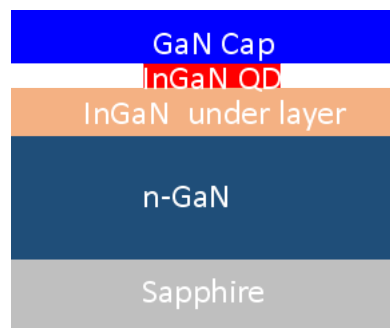
InGaN QD

after PEC etch



Photoluminescence from fabricated InGaN QDs

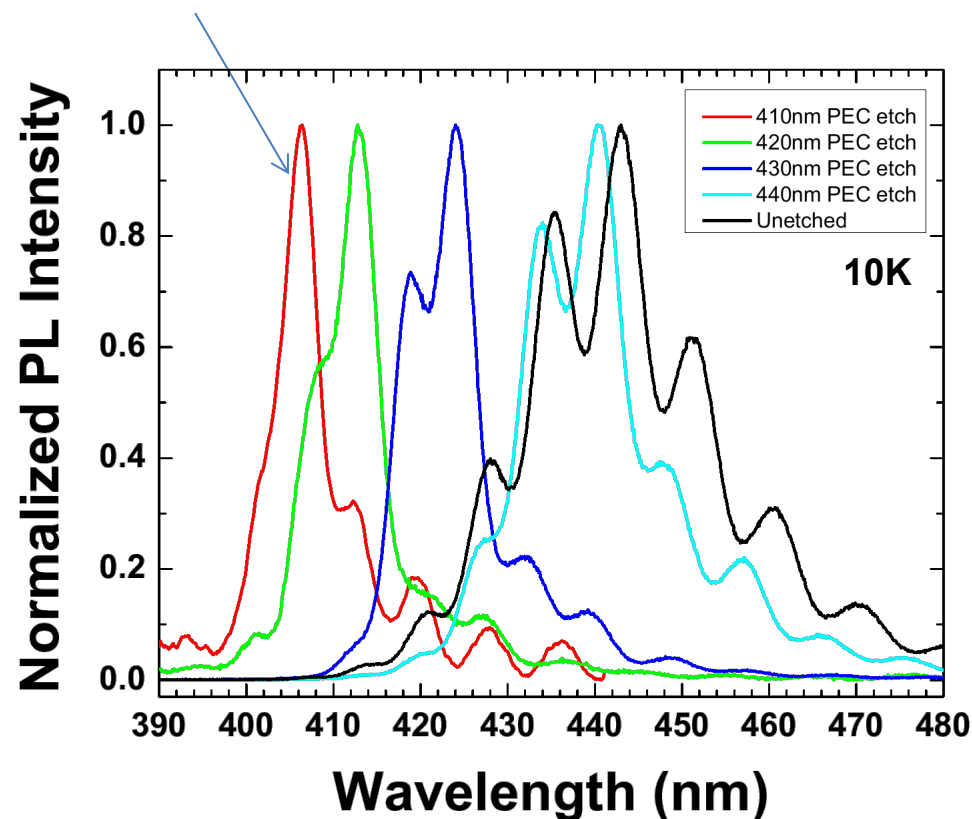
Capped InGaN QW



Photoluminescence (PL) data:

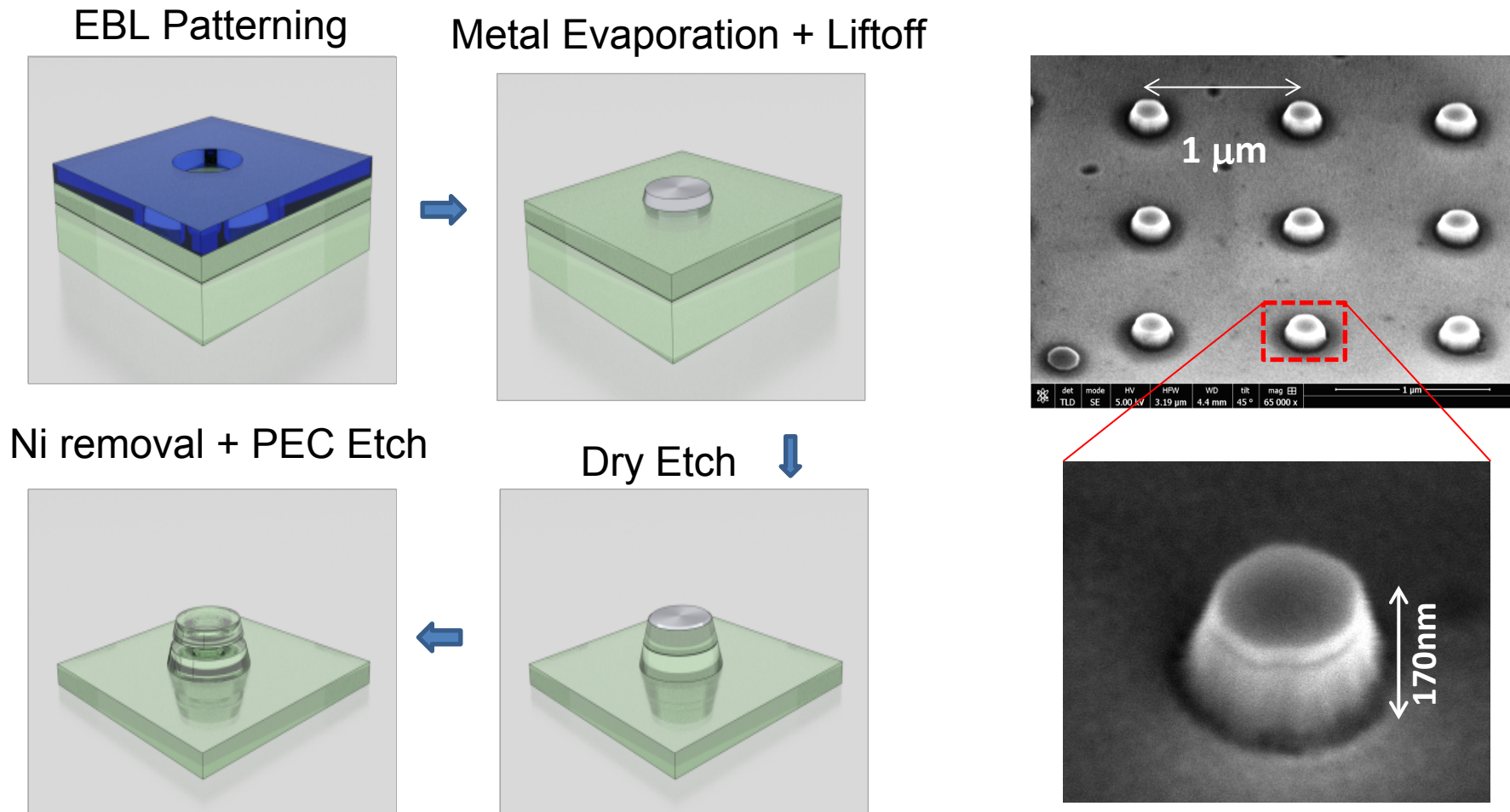
- 375 nm pump (ps pulsed)
- 10K PL data
- PL wavelength determined by PEC etch wavelength
- PL linewidth: 24 nm → 6 nm
- **Quantum size-controlled PEC etching works!**

As narrow as **6 nm** FWHM is consistent with a narrowing of the QD size distribution



Patterned fabrication of InGaN QDs

➤ Deterministic placement



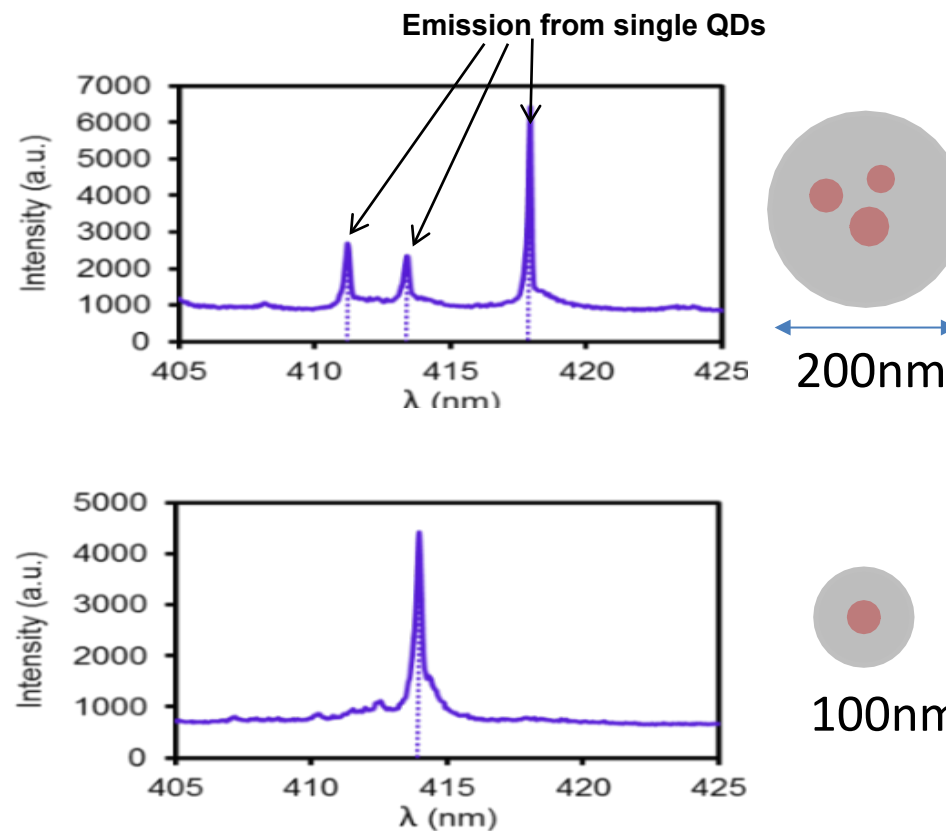
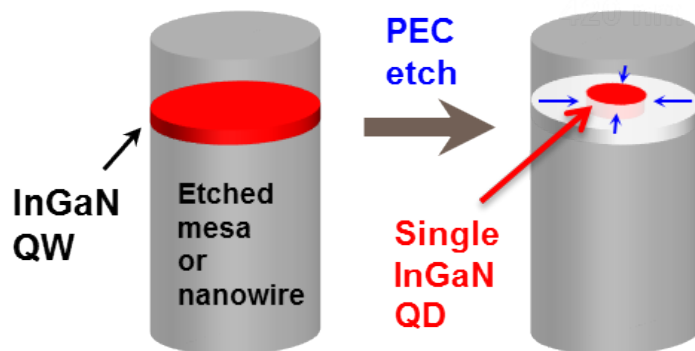
PL from InGaN QDs

Capped InGaN QW



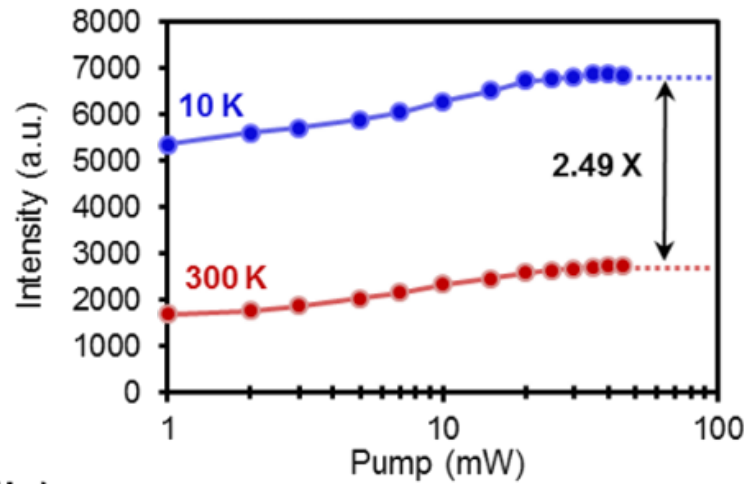
- Posts (100 – 200 nm) patterned with e-beam lithography
- Thicker GaN capping layer (~ 30 nm)
- Narrow PL emission (<1 nm FWHM) observed
- Better ratio of single QD mission to background

Fabrication of single InGaN QDs (methodology)

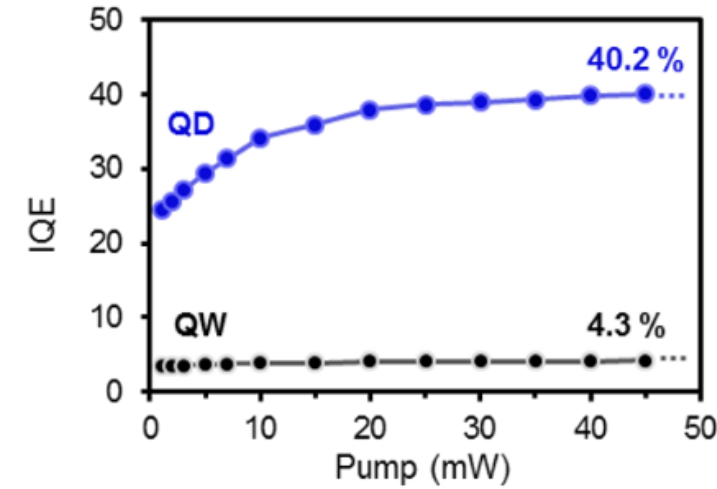


IQE from InGaN QW vs QDs

QD PL pump dependence



IQE QD vs QW

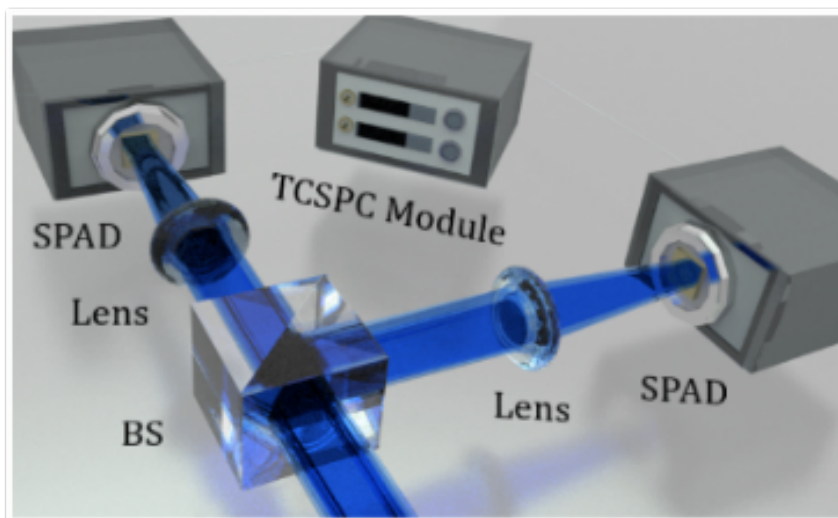


- Significant improvement in IQE between QW and QDs
- Cause: QDs are efficient emitters ; Also poor material etched away

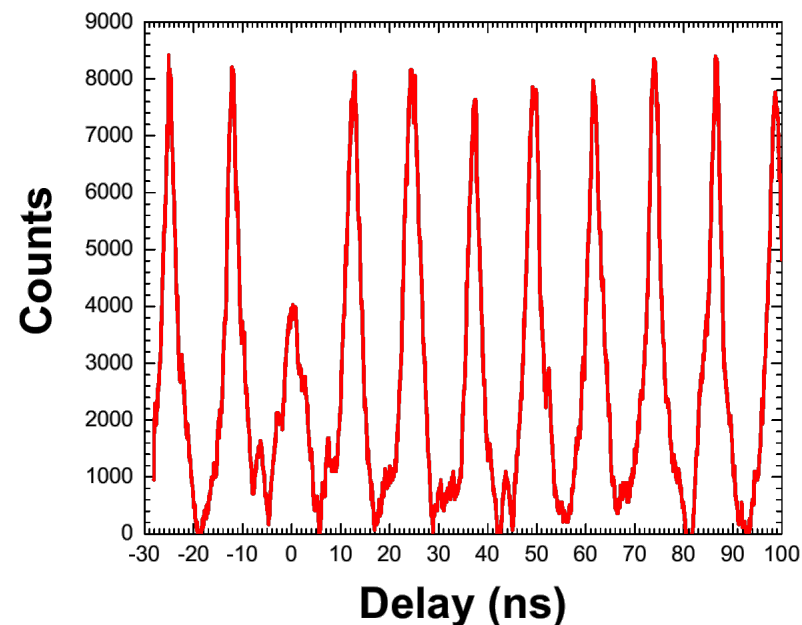
Single Photon Measurement: HBT experiment

- Observation of antibunching : $g_2 \sim 0.5$ for PEC-etched InGaN QDs
- Ways to improve:
 - Thicker GaN cap
 - Reduce nanowire radius
 - Emission rate and light extraction enhancement (Incorporation into PhC)

HBT Experimental Set-up

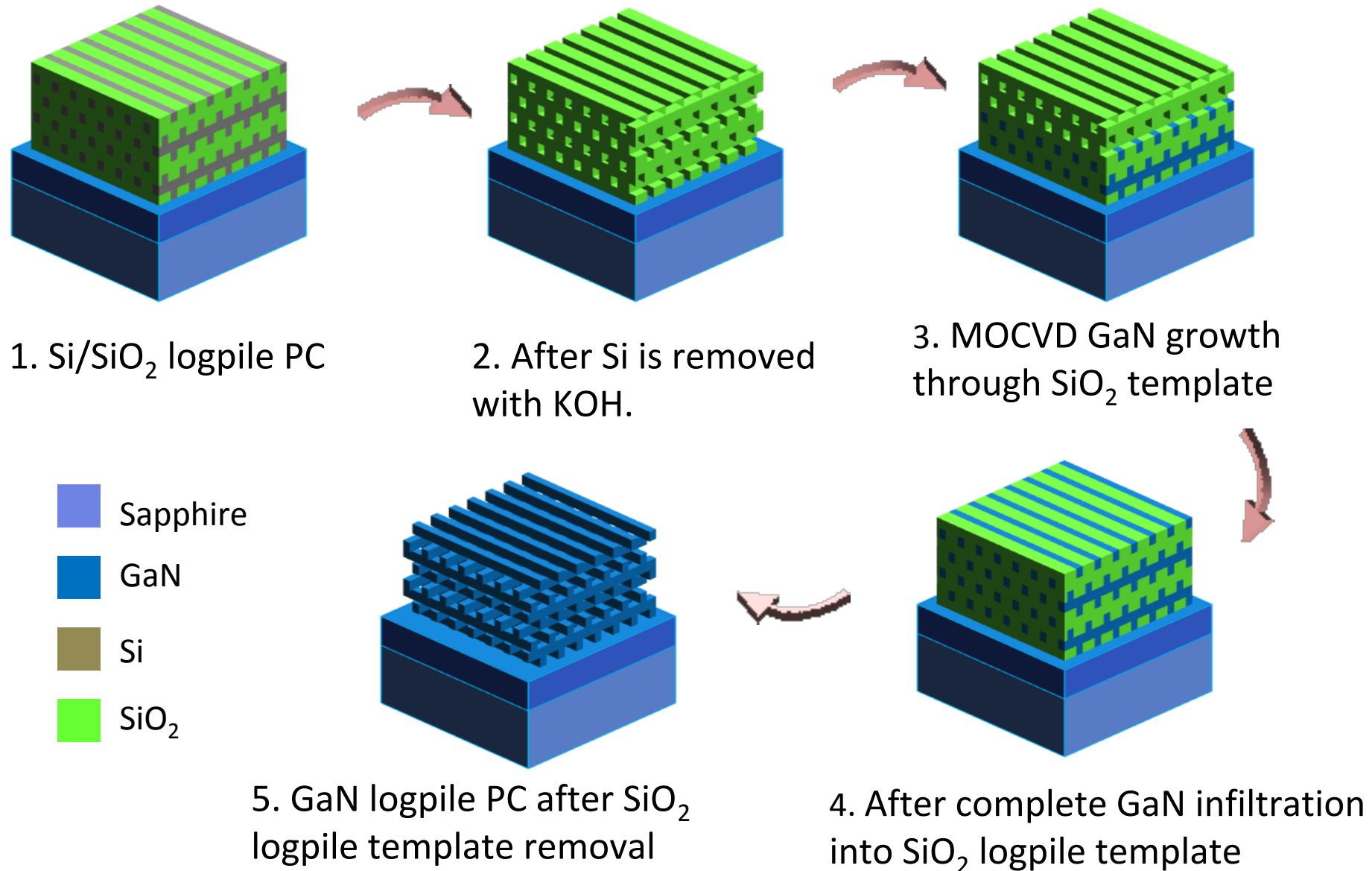


Pulsed g_2 data

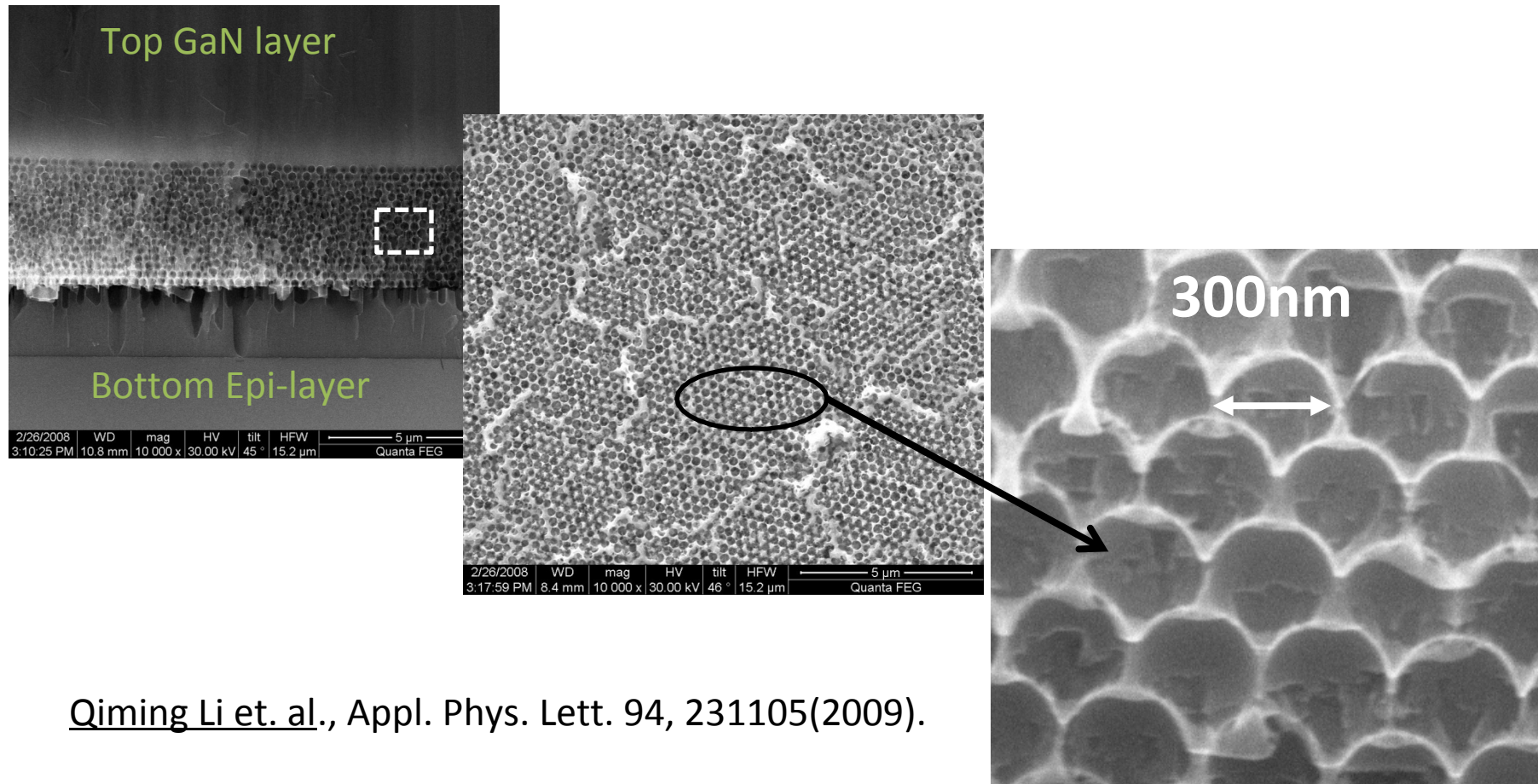


GaN 3D Photonic crystals

GaN growth through logpile template

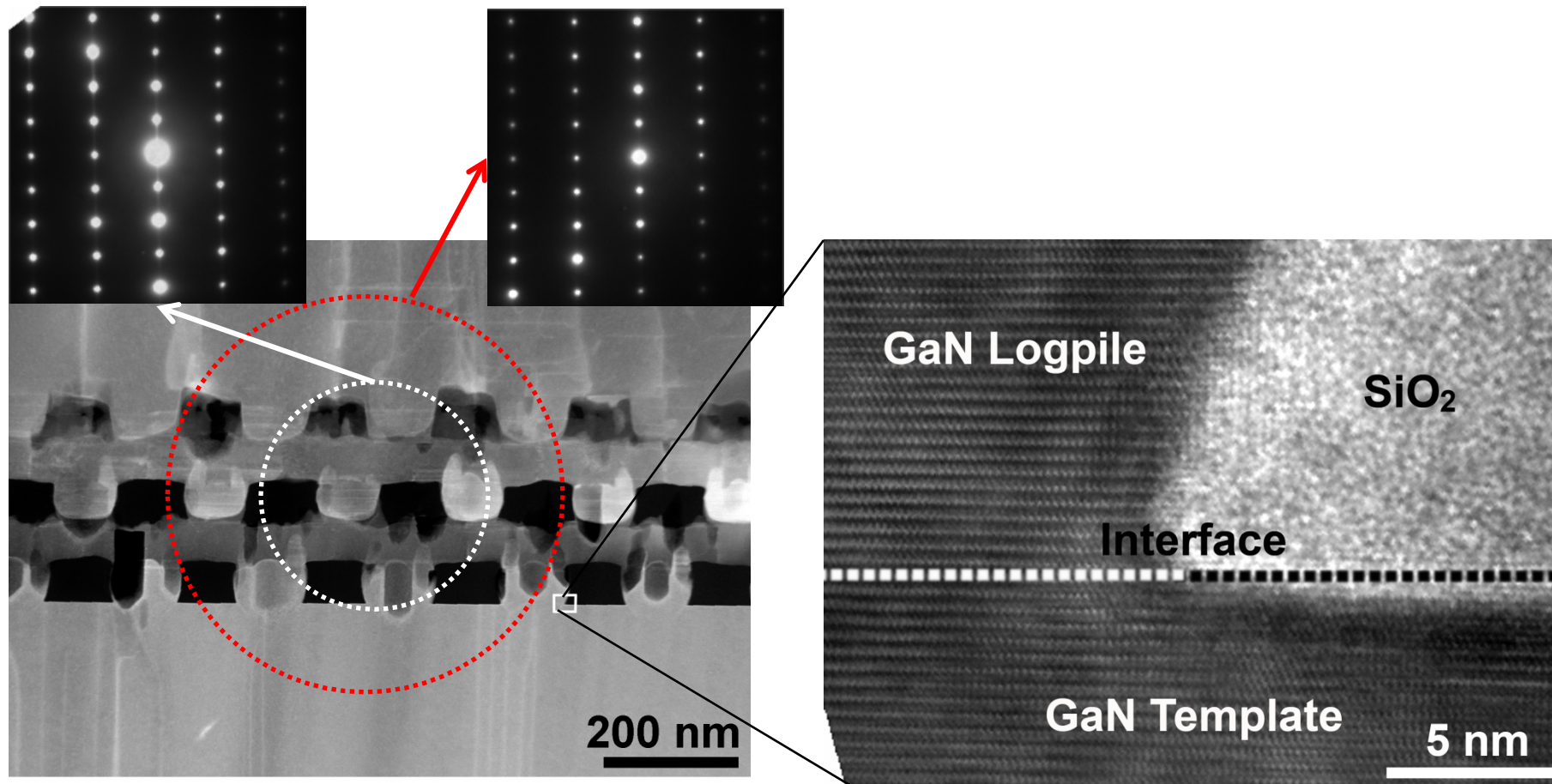


Epitaxial GaN grown through nanostructured template reduces dislocation density



Qiming Li et. al., Appl. Phys. Lett. 94, 231105(2009).

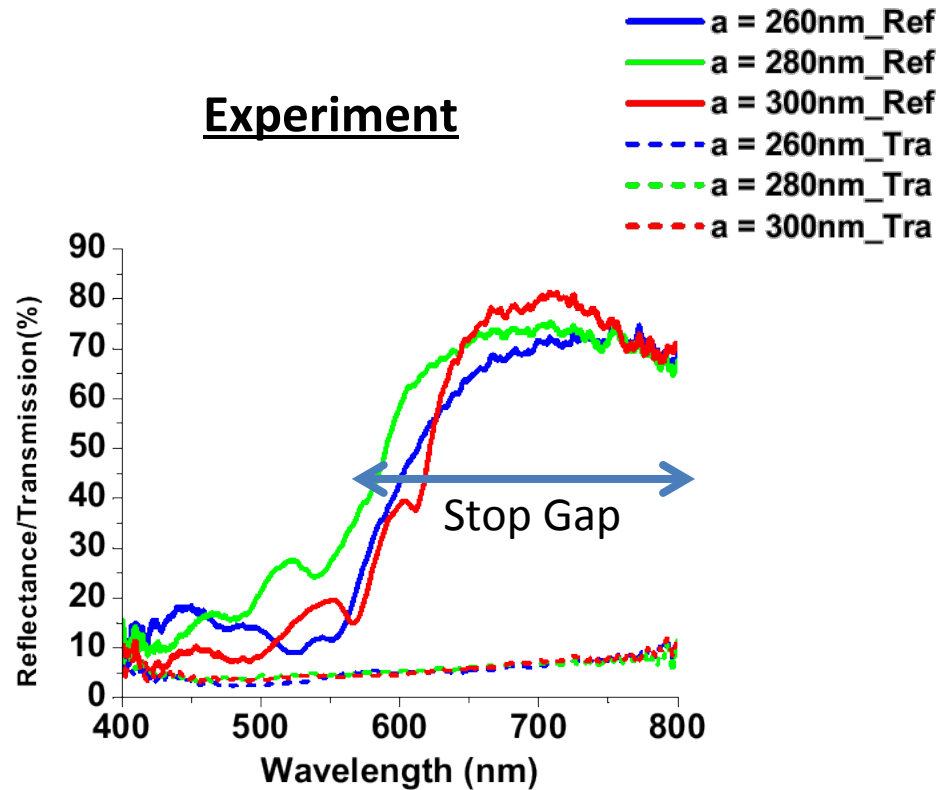
Bottom-up epitaxial growth of crystalline GaN



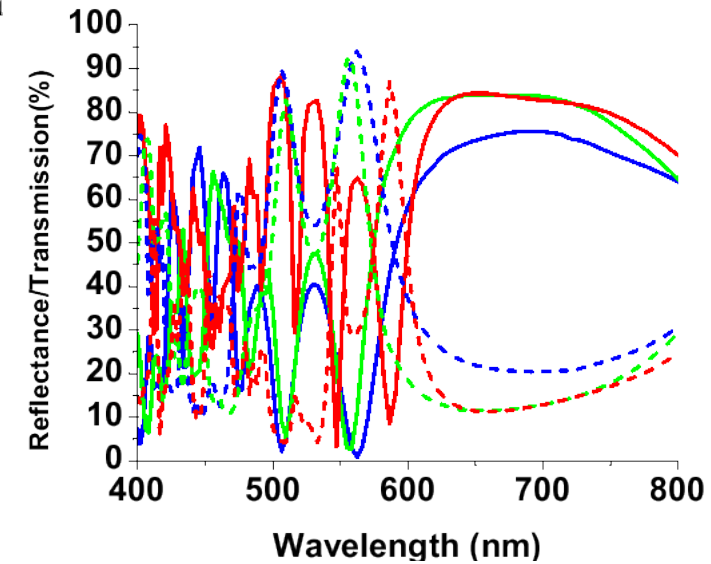
G. Subramania et. al. , Nano Lett. **11**, 4591 (2011).

Optical Response

Experiment



FDTD Simulation

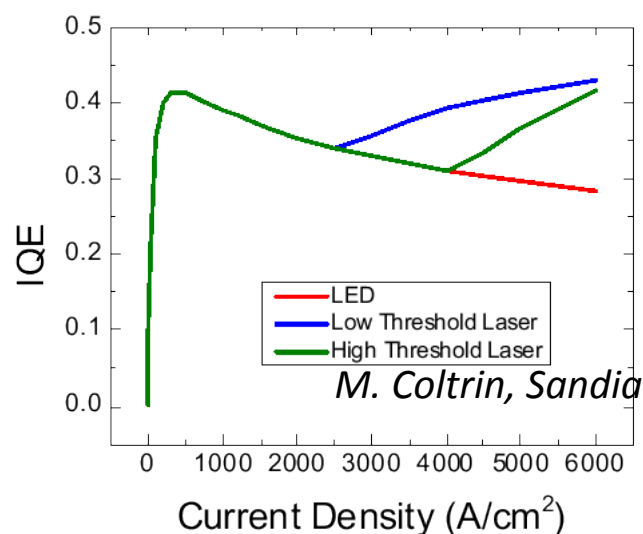


G. Subramania et. al. , Nano Lett. **11**, 4591 (2011).

Nanowire Photonic Crystal Lasers

(Jeremy B. Wright, Sheng Liu, Alexander Benz, George T. Wang, Qiming Li, Daniel D. Koleske, Ping Lu, Huiwen Xu, Luke Lester, Ting S. Luk, Igal Brener)

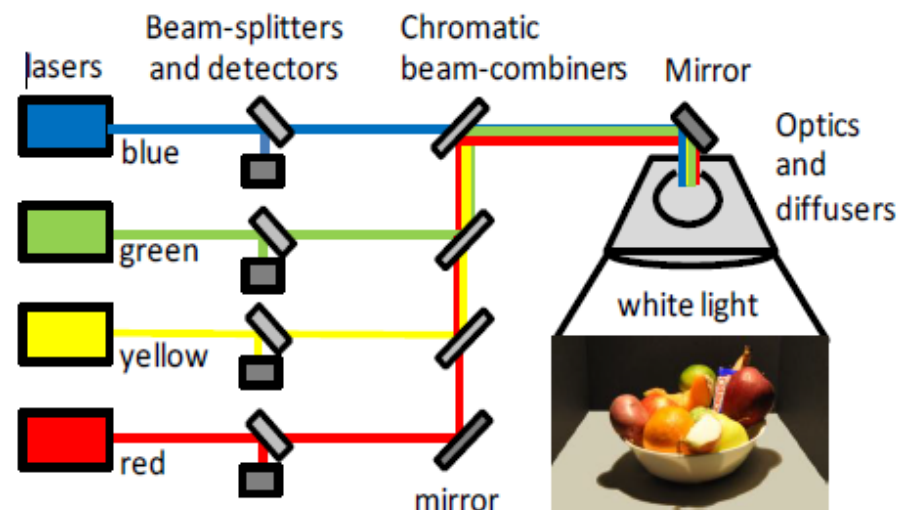
Why Lasers in Solid State Lighting?



The LED “efficiency droop” problem:

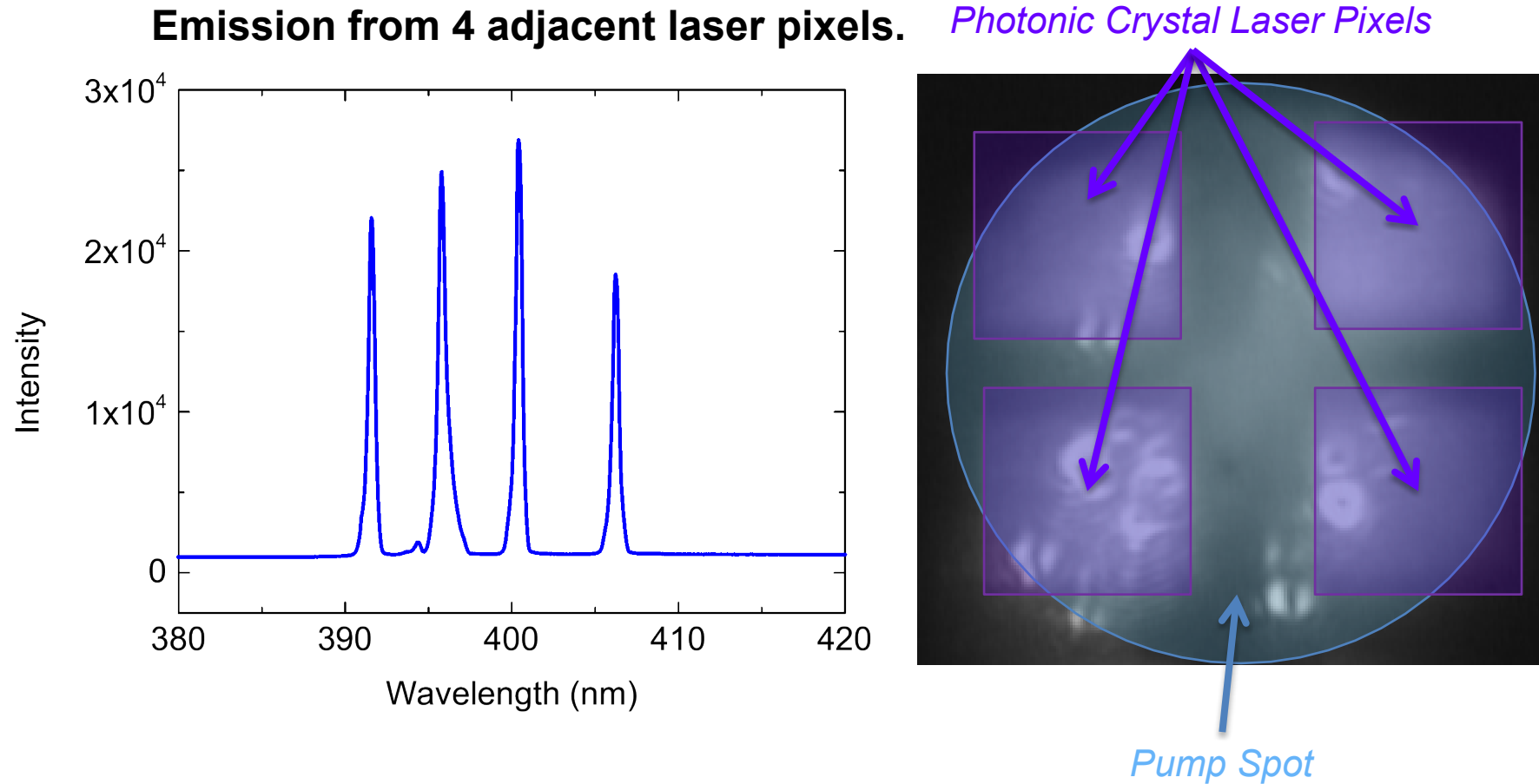
- *InGaN LEDs exhibit a decrease in efficiency at high drive currents*
- *limits the operating current density*
- *increases the cost per lumen*

- *4 color lasers suitably combined can achieve high color rendering*
- *Lasers can potentially mitigate the efficiency droop problem !*



A. Neumann et al., "Four-color laser white illuminant demonstrating high color-rendering quality," *Opt. Express*, vol. 19, pp. A982-A990, 2011.

Multiple Laser Pixels Coexcited



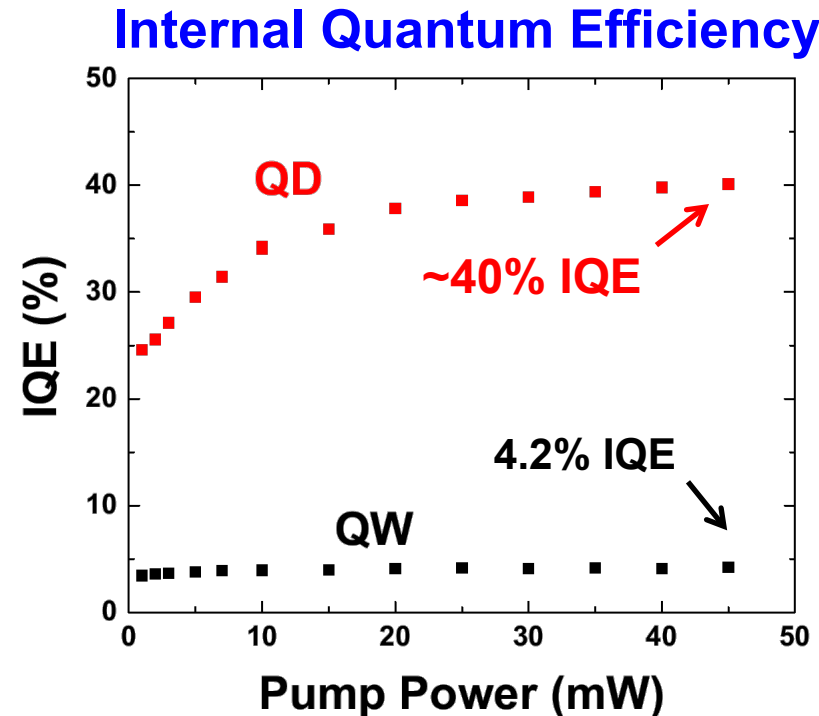
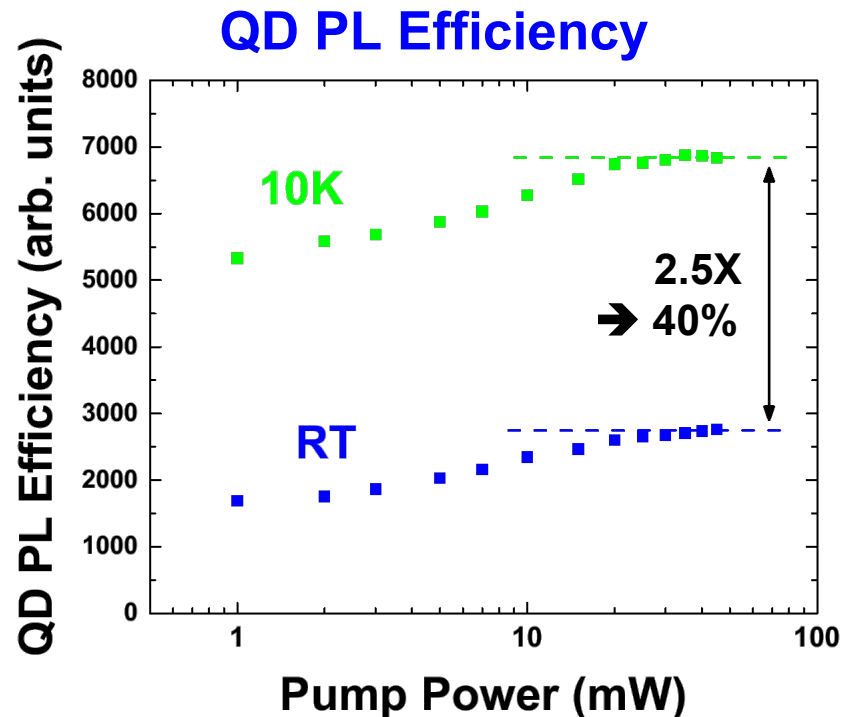
Wright, J. B. *et al.* Multi-Colour Nanowire Photonic Crystal Laser Pixels. *Sci. Rep.* **3**, 2982, doi:10.1038/srep02982

InGaN QD internal quantum efficiency

Capped InGaN QW

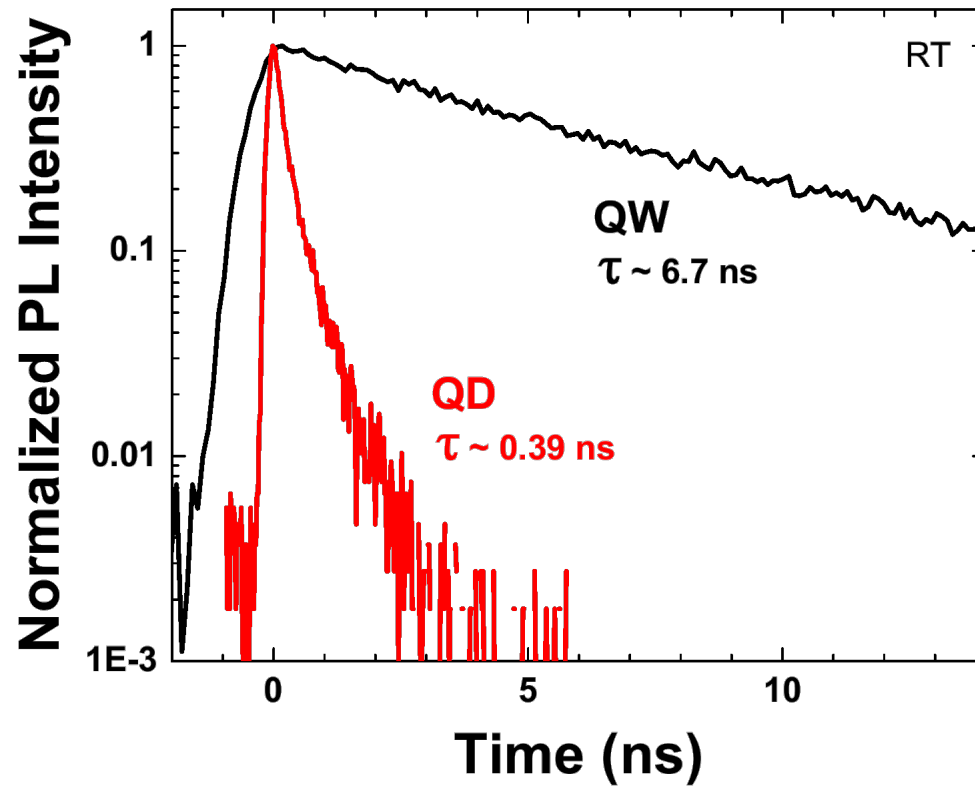


- Compare 10K and RT PL efficiency
- Assumes 10K PL is 100% efficient
- PL Intensity drops by >100X after QD etching
- IQE goes up by almost 10X after QD etching
- QDs are expected to have better IQE



Time-resolved PL data from InGaN QDs

Capped InGaN QW

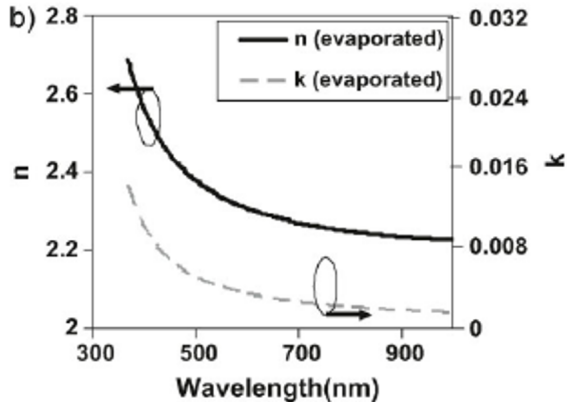


TRPL data:

- 405 nm pump (~ 2 ps, pulsed)
- Resonant pumping into InGaN
- Room temperature TRPL data
- Hamamatsu streak camera data
- 17X change in PL lifetime
- Lifetime is expected to be much shorter for QDs
- Shows that we have fundamentally changed the InGaN material
- QW $\rightarrow$ QDs

Visible Frequency Logpile Photonic Crystal

Choice : TiO_2



Requirements:

- High enough index with low loss in the visible :
 - $n \sim 2.3-2.4$ in the visible /w $k < 0.015$
- Easy to deposit :
 - Sputtering, evaporation, solgel, Atomic Layer Deposition

Band Structure Calculation

Solve Electromagnetic wave equation for Plane wave

$$\begin{cases} E(\mathbf{r}, t) \propto e^{-i\omega t} \\ H(\mathbf{r}, t) \propto e^{-i\omega t} \end{cases}$$

$$\nabla \times \left(\frac{1}{\epsilon(\mathbf{r})} \nabla \times \mathbf{H}(\mathbf{r}) \right) = \left(\frac{\omega}{c} \right)^2 \mathbf{H}(\mathbf{r})$$

In a periodic potential

$$\epsilon(\mathbf{r} + \mathbf{R}) = \epsilon(\mathbf{r})$$

$$\mathbf{R} = a_1 \hat{x} + a_2 \hat{y} + a_3 \hat{z}$$

Apply Bloch's Theorem

$$\mathbf{E}, \mathbf{H}(\mathbf{r} + \mathbf{R}) = \mathbf{E}, \mathbf{H}(\mathbf{r})$$

Plane Wave Expansion Method

Reciprocal (K) space

Reciprocal Lattice vector

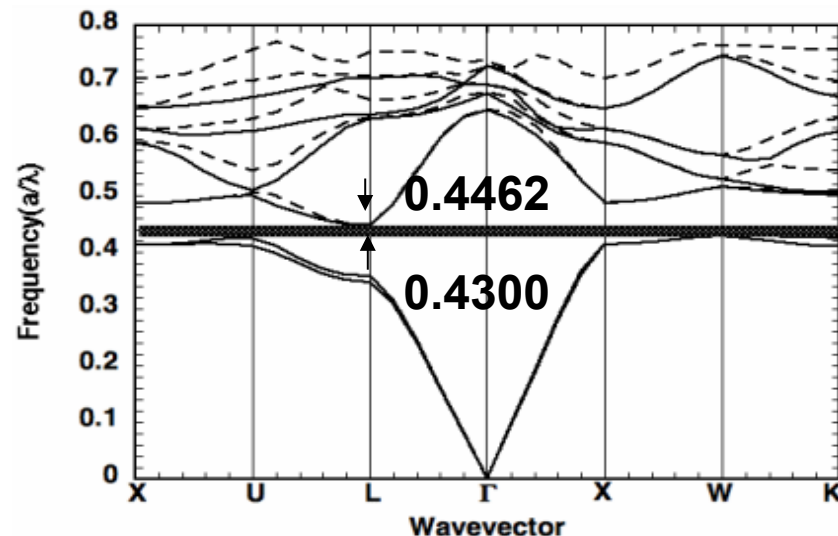
$$\varepsilon(\mathbf{r}) = \sum_{\mathbf{G}} \bar{\varepsilon}(\mathbf{G}) e^{i\mathbf{G} \cdot \mathbf{r}} \quad E, H(\mathbf{r}) = \sum_{\mathbf{G}} \bar{E}, \bar{H}(\mathbf{G}) e^{-i(-\mathbf{k} + \mathbf{G}) \cdot \mathbf{r}} \quad \vec{G} \cdot \vec{R} = 2\pi n$$

Bloch Waves $\bar{H}(\mathbf{G}) = H_x(\mathbf{G})\hat{x} + H_y(\mathbf{G})\hat{y} + H_z(\mathbf{G})\hat{z}$

Eigen value equation

$$\nabla \times \left(\frac{1}{\varepsilon(\mathbf{r})} \nabla \times H(\mathbf{r}) \right) = \left(\frac{\omega}{c} \right)^2 H(\mathbf{r}) \quad \longrightarrow \quad \boxed{A \bar{H}_n(\mathbf{k}) = -\frac{a^2 \omega_n^2}{c^2} \bar{H}_n(\mathbf{k})}$$

Photonic Band Structure for TiO₂ logpile



1st BZ

U
W

Band index

Wave vector within the first BZ

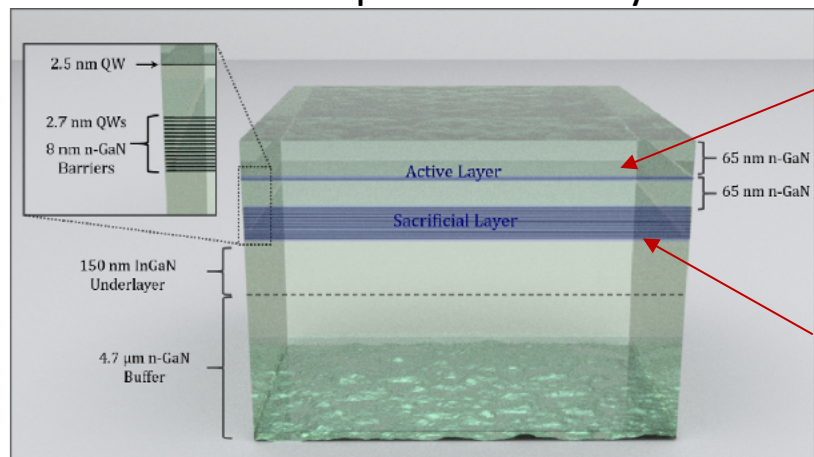
Eigen value equation becomes a matrix equation for which you can write a code and solve for band structure within the first Brillouin zone.

- $n \sim 2.3$
- 4% bandgap
- $\sim 20\text{nm @ } 500\text{nm}$

Example: InGaN/GaN based membrane Photonic Crystal

- Photoelectrochemical (PEC) etching of a sacrificial InGaN layer

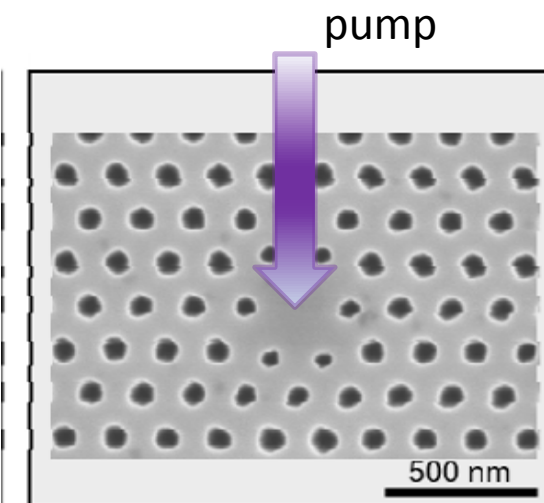
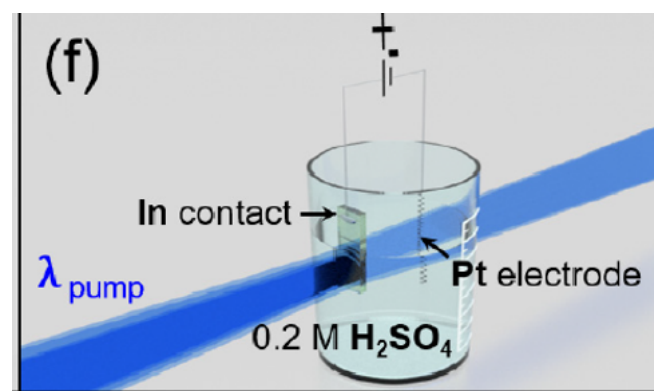
Epitaxial film layout



InGaN SQW
low indium

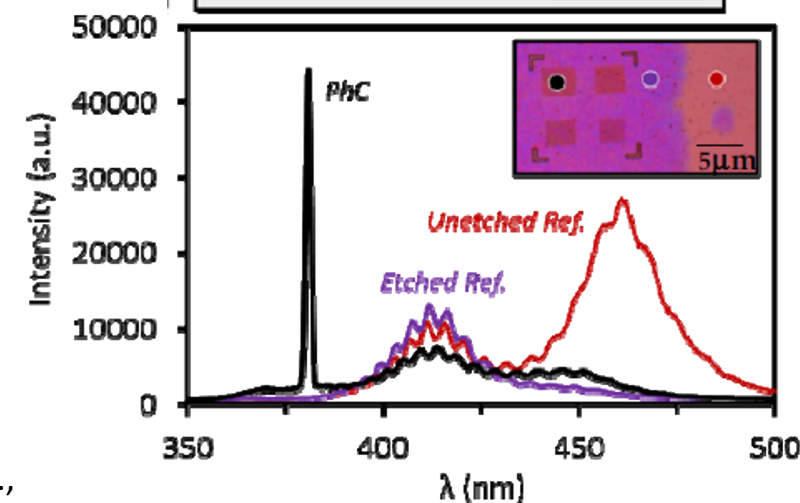
InGaN MQW
higher indium
sacrificial layer

Photoelectrochemical etch

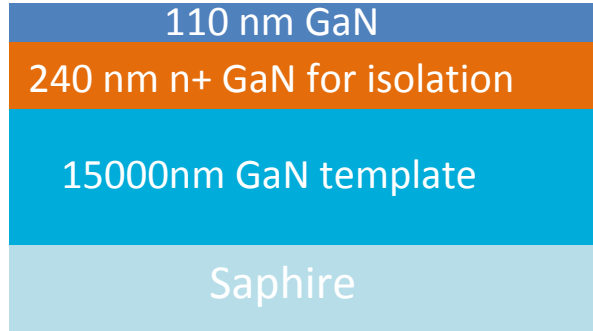


- The epitaxial structure contains a low *In* concentration active layer for PhC membrane
- Higher *In* concentration InGaN/GaN MQW is grown between the bottom substrate
- InGaN layer is etched away using PEC to create a low index layer

Approach is challenging and harder for thicker MQW layer

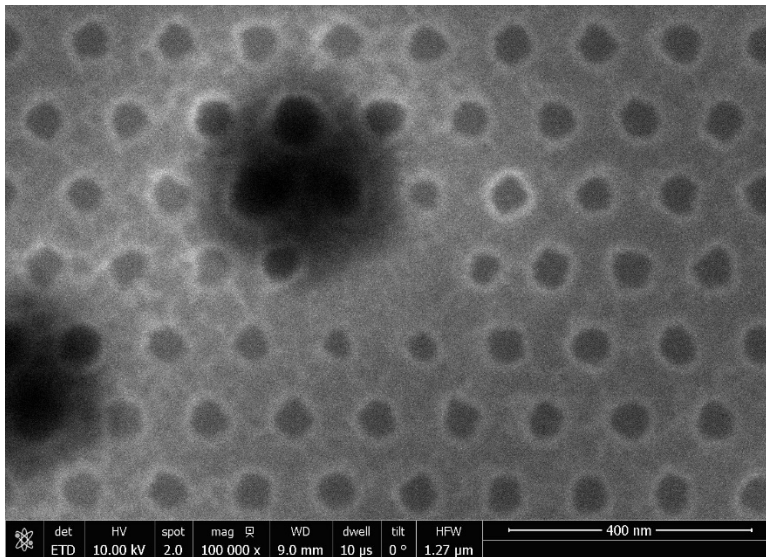


PhC Membrane Fabrication with EC Etch

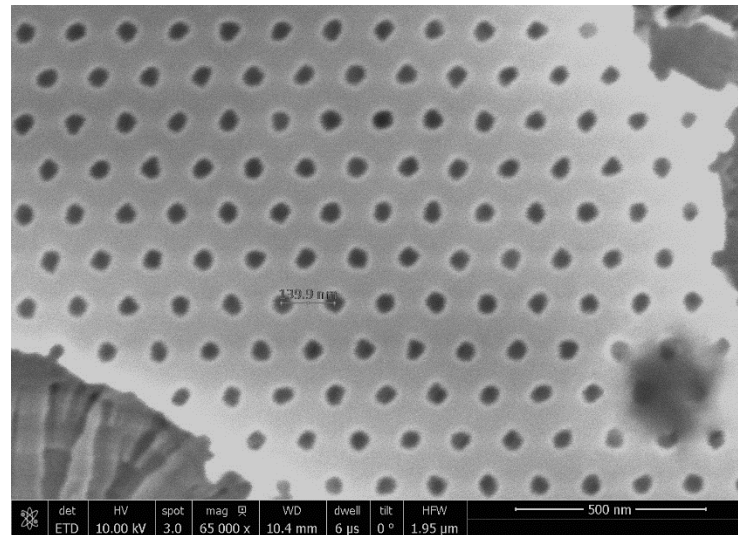


- Follow similar fabrication procedure to form PhC like before till cladding removal step
- EC etch highly doped cladding region

Pre –EC Etch



Post –EC Etch



EC etched region revealing GaN template

