



Structural, Environmental and Mechanochemical Interactions in Solid Lubricant Material Systems



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Sandia National Laboratories, Albuquerque, NM

JAST Webinar

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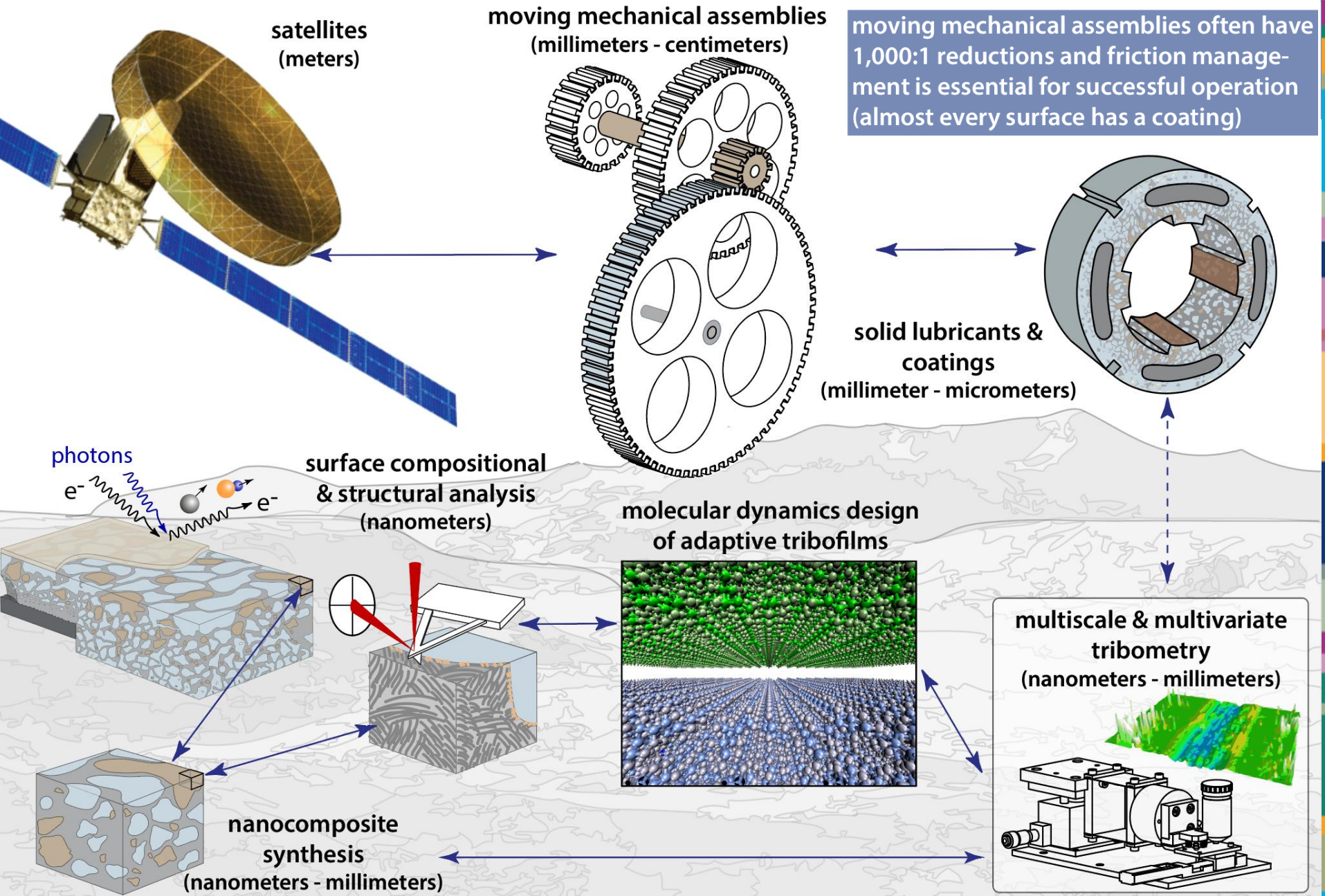


1. Brief Introduction & Motivation

2. Current & Ongoing Research Projects

- Environmental Factors Governing Run-In Behaviors and Oxidation response in MoS₂
 - Aging of Pure/Composite films with Varying Structures
 - ALD MoS₂ Conversion
 - Fundamental Role of Structure on Friction Response
- Understanding Variability in Run-in for DLC Coatings
 - High Throughput Assessment of Different Film Compositions
 - Deposition Directed Surface Termination Studies
- Self-lubricating in situ carbon films (SLIC)

Fundamental Studies & Applied Challenges



Extreme Operating Environments



Space:

- operate in vacuum (atomic oxygen in low earth orbit)
- store months – years before use; generally non-serviceable
- operating temperatures from 50 – 300K, depending on location
- large investments of time and money



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Precision Mechanisms:

- inert gas near P_{atm} , trace O_2 , H_2O , outgassing species
- store for decades; non-serviceable
- operating temperatures 200 – 350K
- large investments of time and money
- consequences (political, societal) of failure are unacceptable



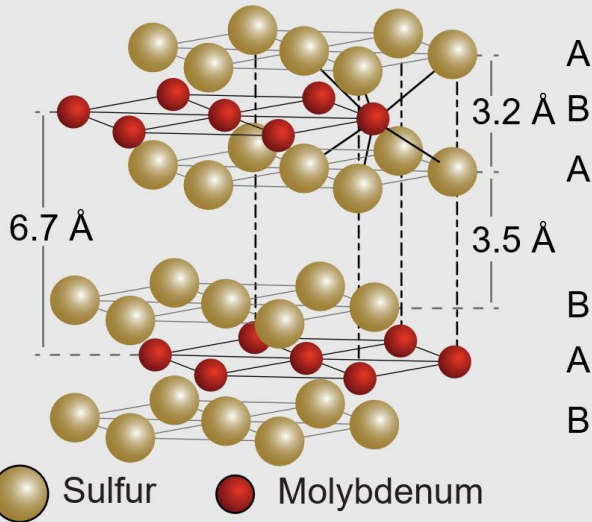
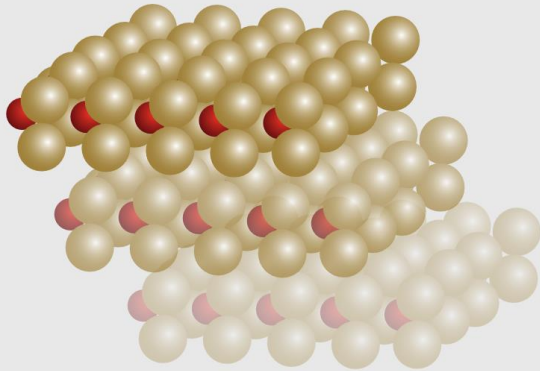


Environmental Factors Governing Run-In Behaviors and Oxidation response in MoS₂

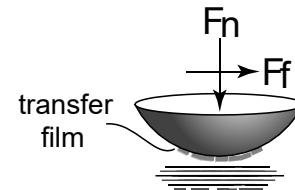
7 MoS₂ – How it Works



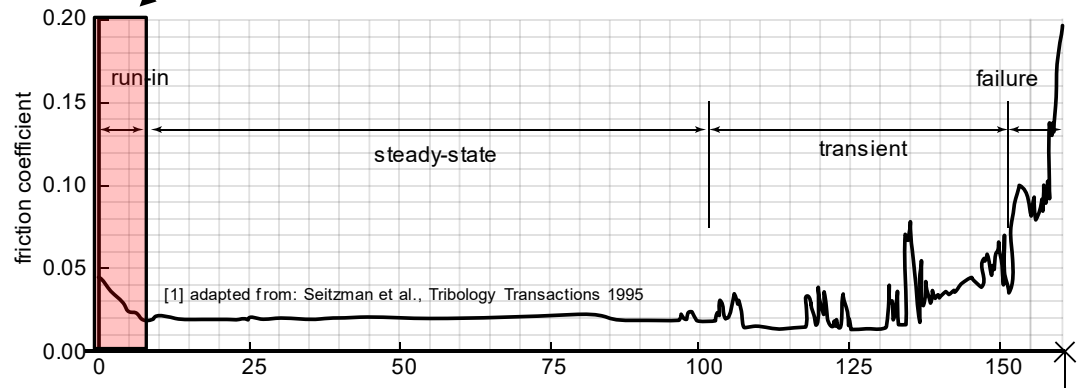
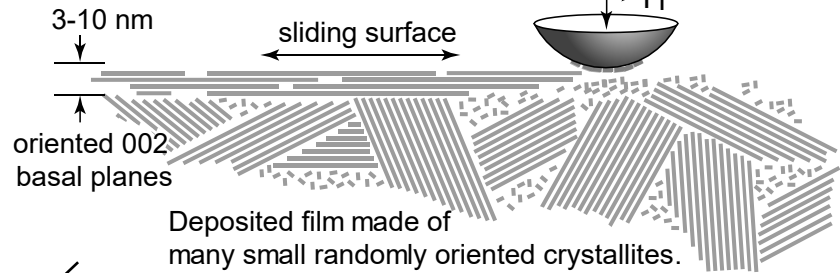
molybdenum disulphide
 $\mu = 0.02 - 0.06$ (inert @ 1N)
 $\mu = 0.15 - 0.25$ (humid air @ 1N)



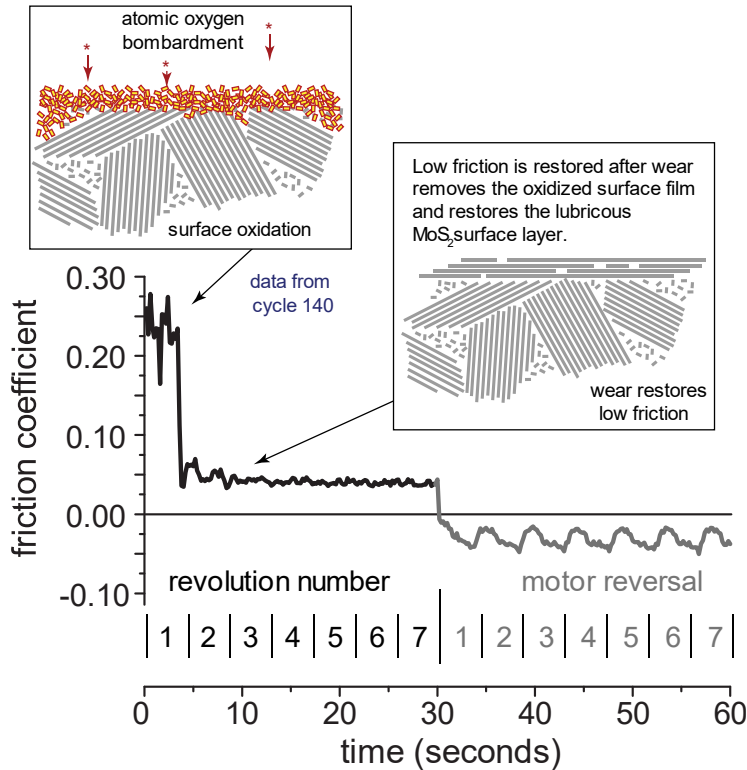
generalized run-in processes



2) Shear-induced
 crystallite re-orientation



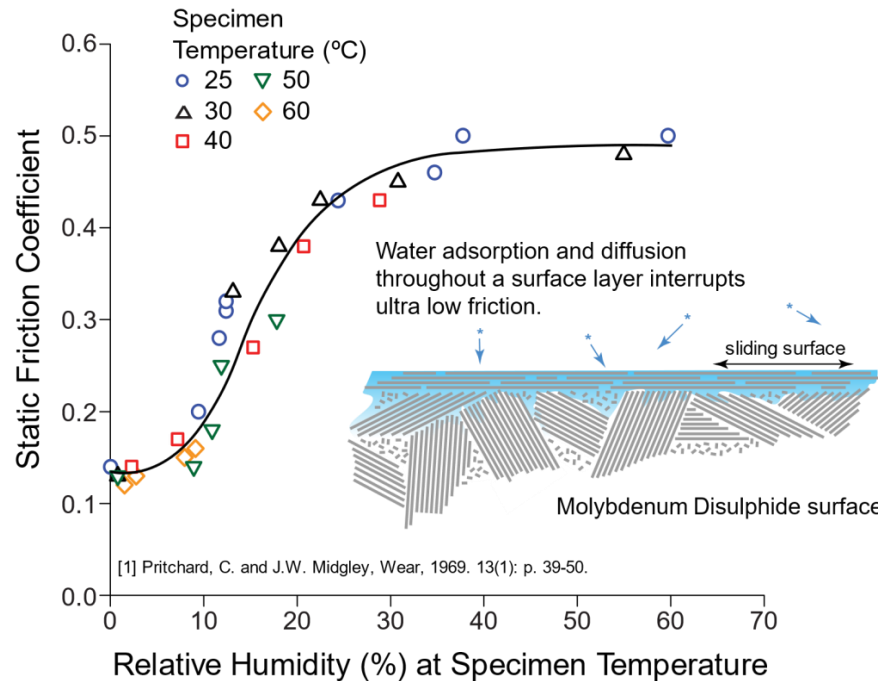
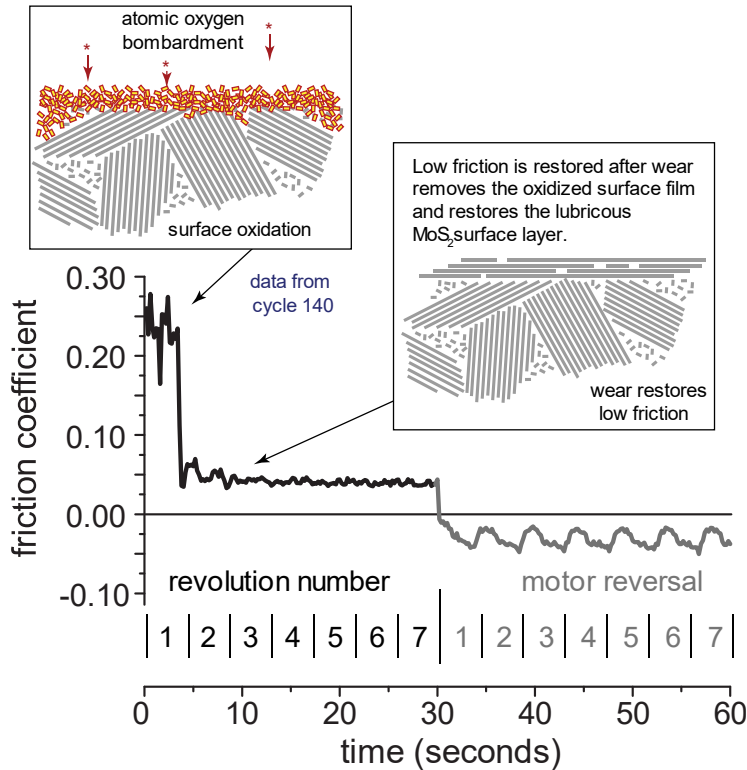
Typical friction trace for MoS₂ lubricated contact. Initial run-in is followed by steady state low friction which ultimately transitions to high friction before failure [1]



Krick et. al, unpublished

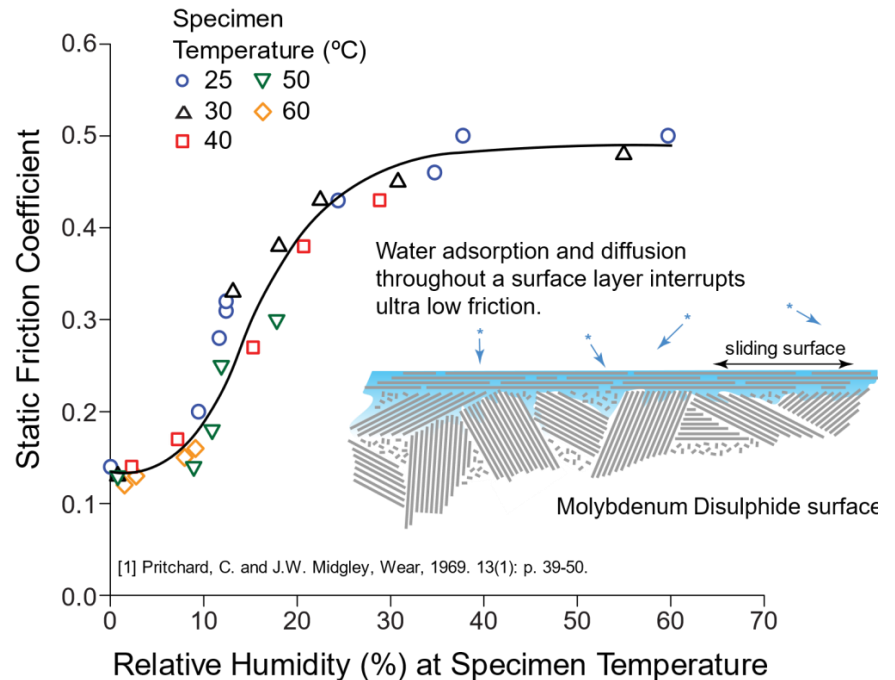
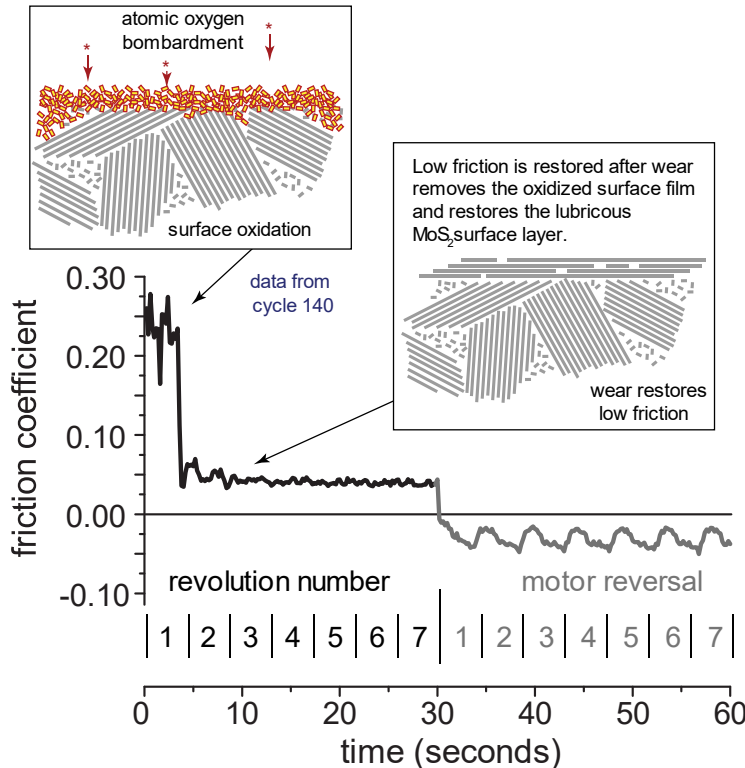
- Oxidation can occur in space (AO - fast), air at high temps (O_2 – fast) and room temp (H_2O – slow)

Bad Actors – Environment & Aging



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- Water enhances static and kinetic friction behaviors via increased shear between layers

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Many components operate infrequently and for very few cycles – **effectively living in the run-in regime**

Initial Friction Evolution (Run-In)



Drawbacks

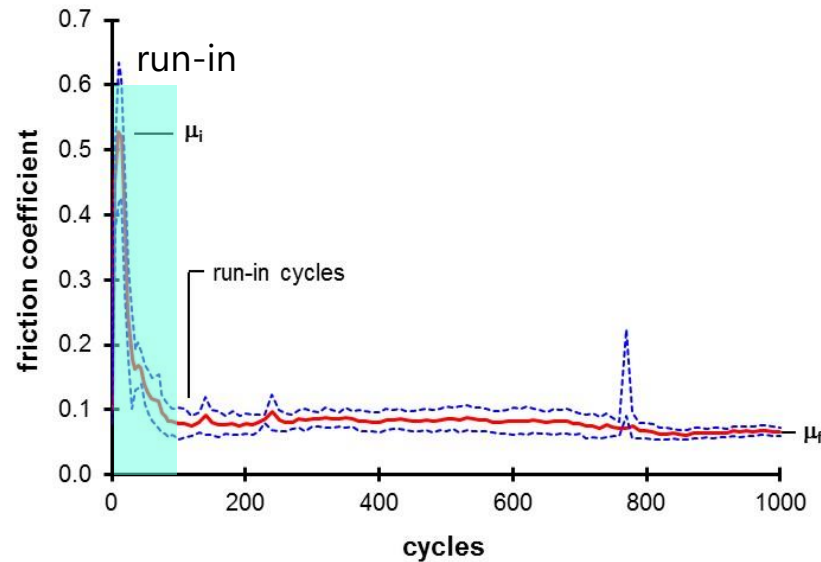
- Increased friction/wear
- Risk of seizure
- Reduced performance margin in mechanisms
- Loss of energy

Causes

- Structural (re-orientation, crystallization)
- Environmental (oxidation, adsorbates)

Solutions

- Tailor composition (compositing for densification, water-getting)
- Tailor microstructure, encourage ordering (impingement films, capping layers, deposition inputs)



Initial Friction Evolution (Run-In)



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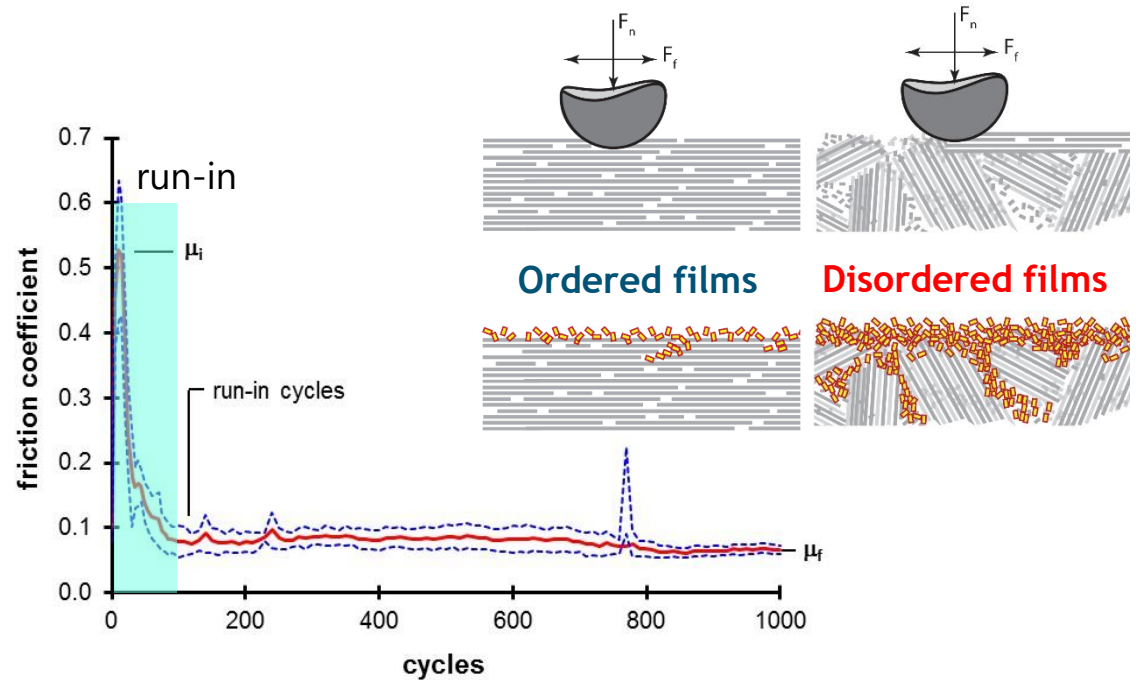
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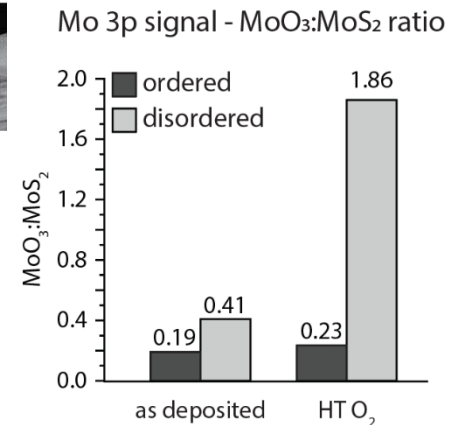
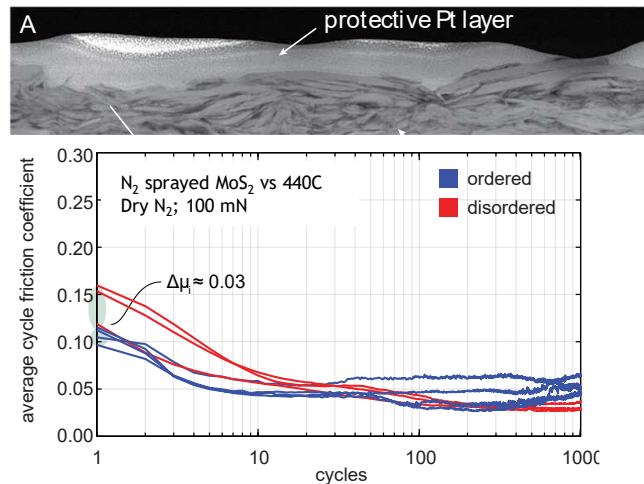
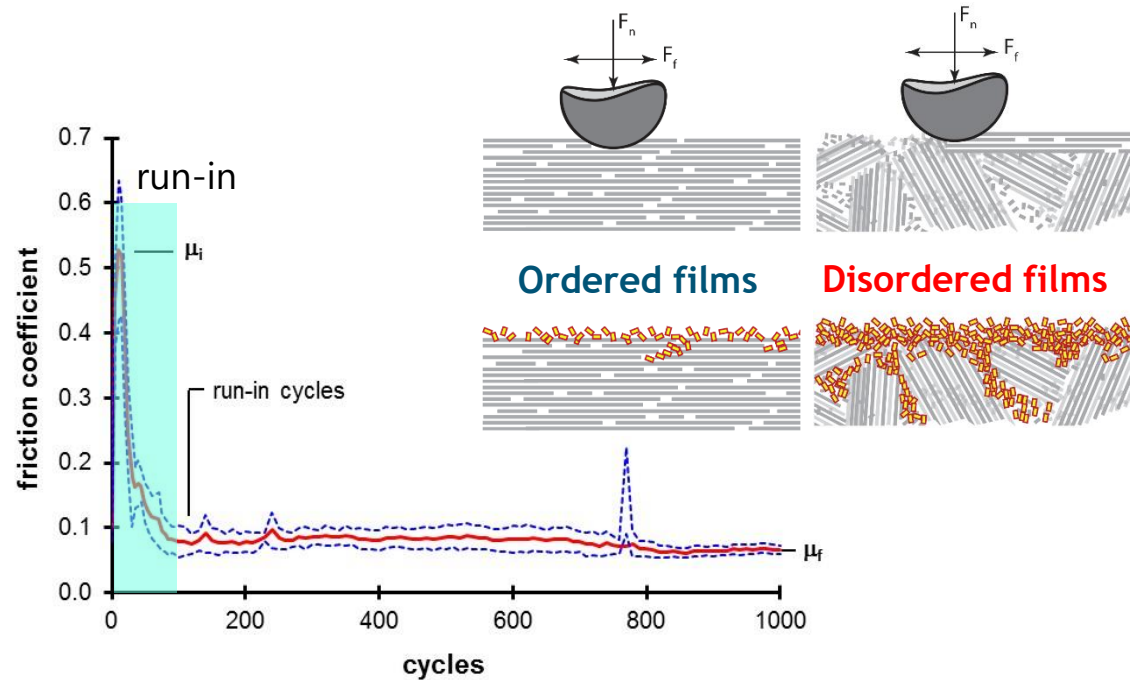
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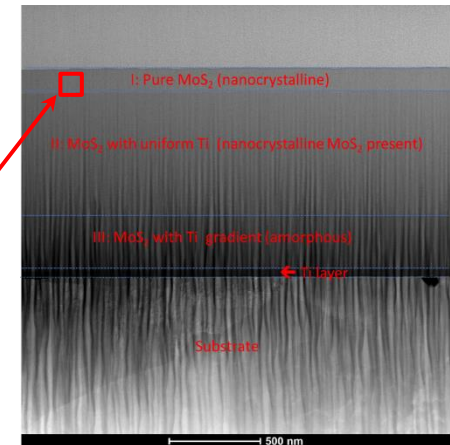
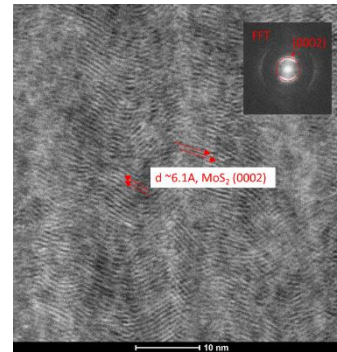
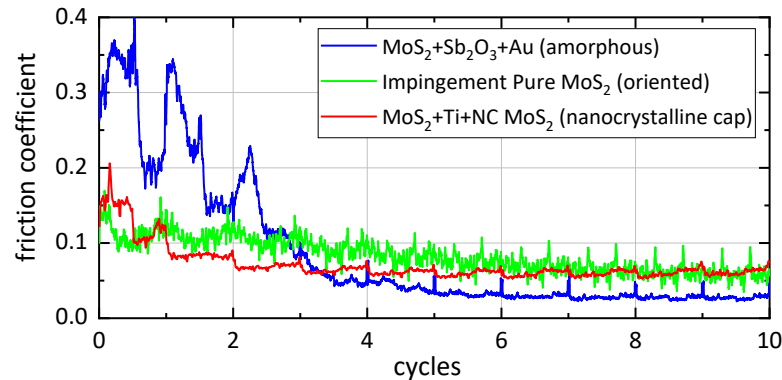


Efforts to Reduce Run-In: Structure/Composition

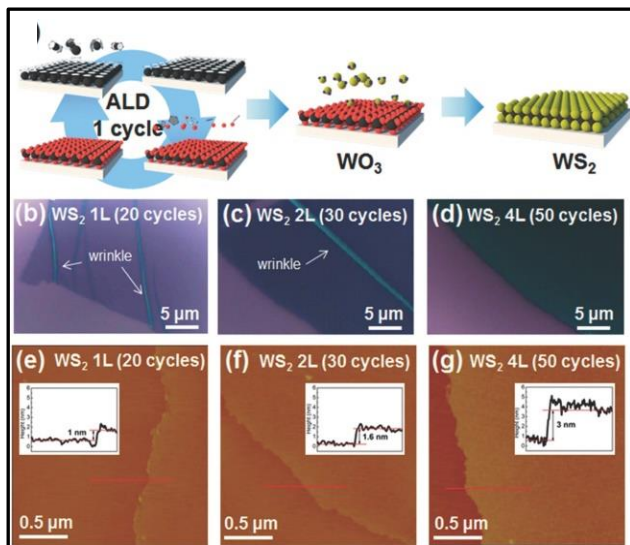


PVD nanocrystalline & composite films

- Films with NC structures exhibit reduced run-in compared to amorphous films (composites); likely reduced reactivity as well



MoS₂+Ti+pure MoS₂ cap



MoS2 ALD at LBNL Molecular Foundry

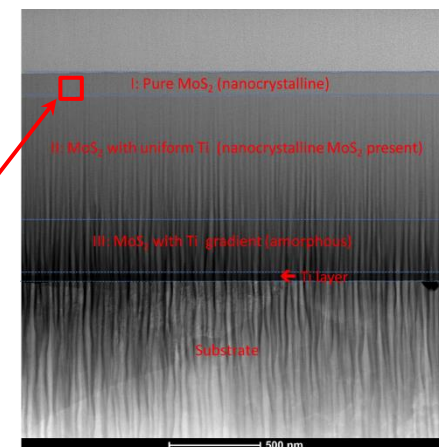
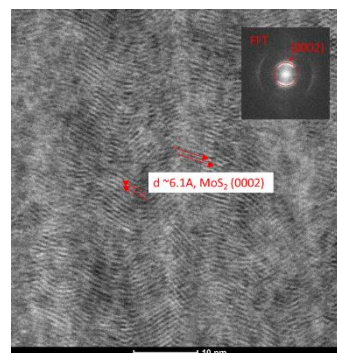
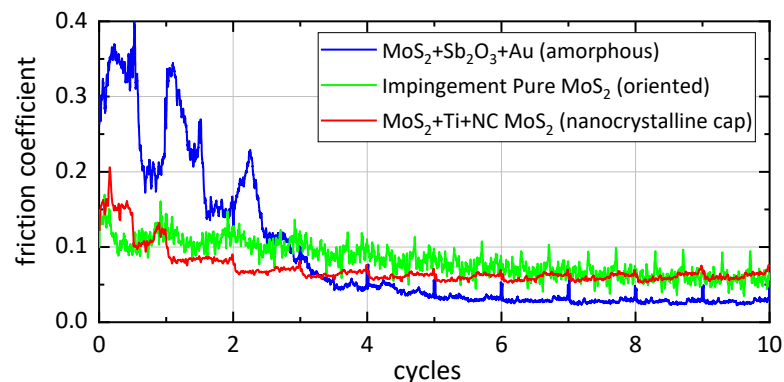
- Developing ALD MoS₂ freestanding and capped films for aging resistance
- MoO_x conversion with Moly hexacarbonyl precursor and H₂S conversion
- Challenges remain in optimizing synthesis to promote full conversion at low enough temperatures

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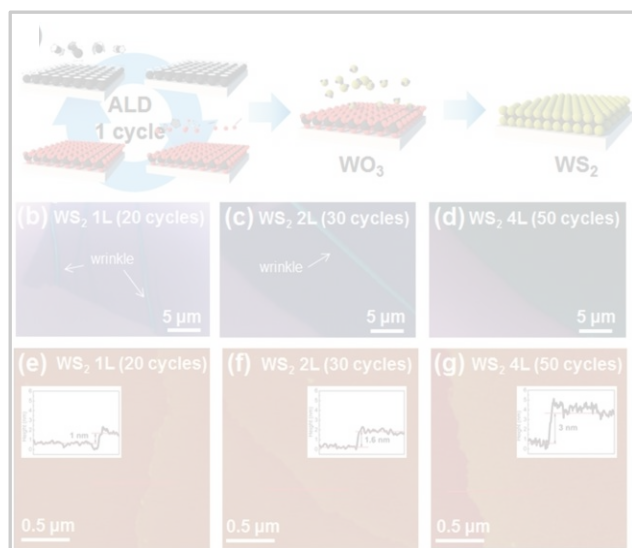


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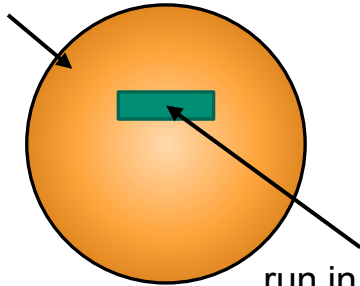
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Run In

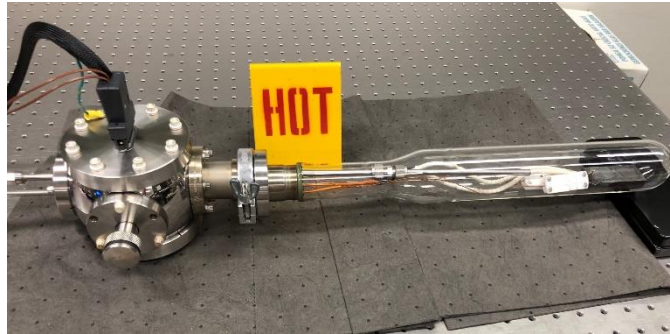
coated disk



run in
patch 4x8
mm

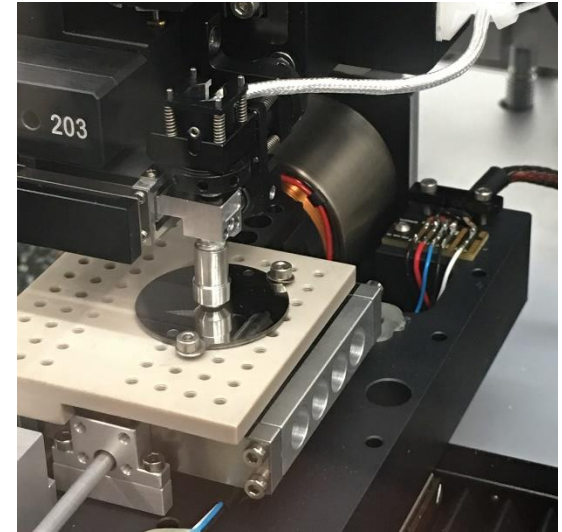
- 13-8PH or 440C stainless steel disks
- run in at 530 MPa, 50 passes, overlapping areas

Accelerated Aging



- 200°C, dry (DP < -60°C) air, 5 SCFH
- 12 hours

Friction Testing

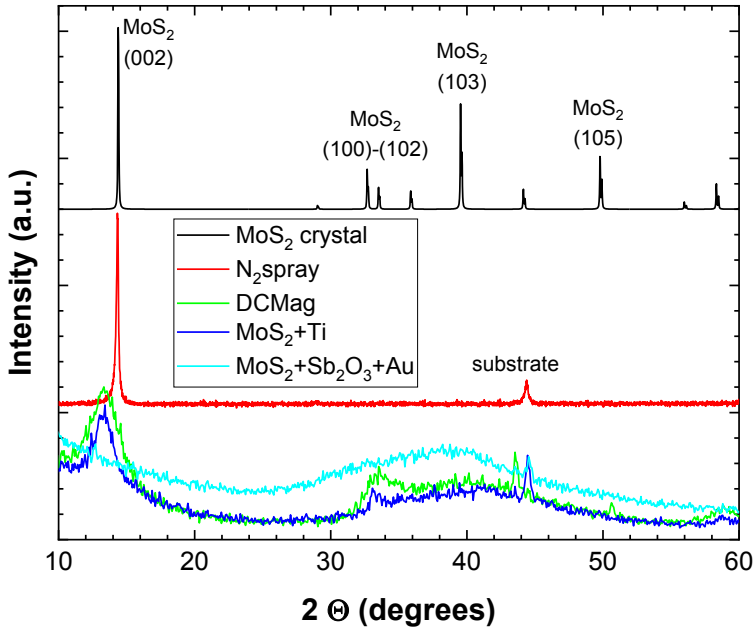


- 440C ball, 3.2 mm dia.
- 1 mm/s sliding speed
- Hertz contact pressures of 275, 530 and 785 MPa

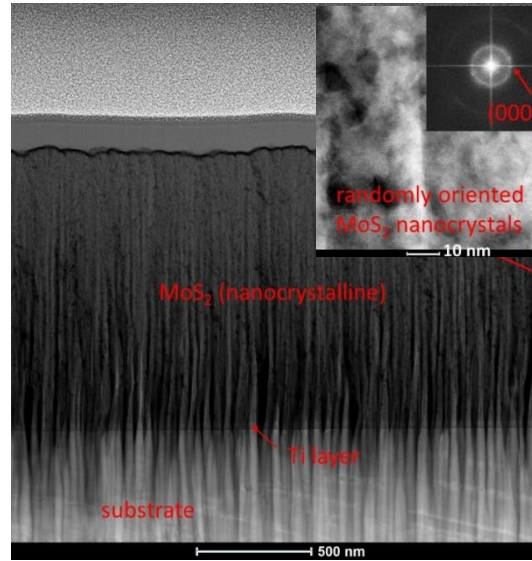
Materials Investigated:

- N₂ (pure MoS₂ sprayed with N₂)
 - DC (pure DC sputtered MoS₂)
 - Ti (RF sputtered MoS₂, Ti-doped)
 - Sb₂O₃/Au (RF sputtered Sb₂O₃+Au-doped MoS₂)
- } **Pure MoS₂**
- } **Doped MoS₂**

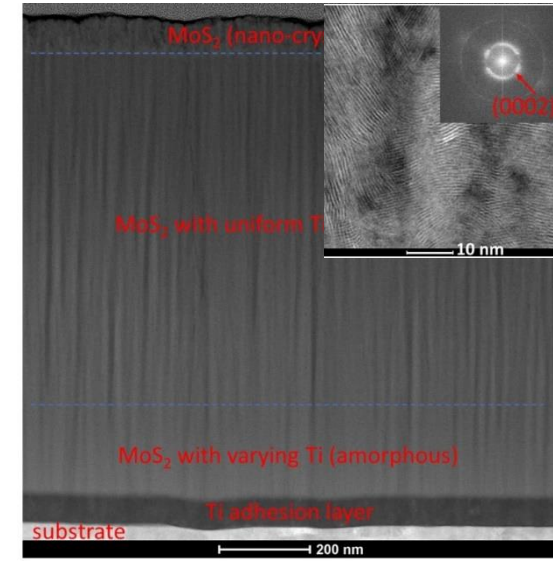
Coating Microstructures & Compositions



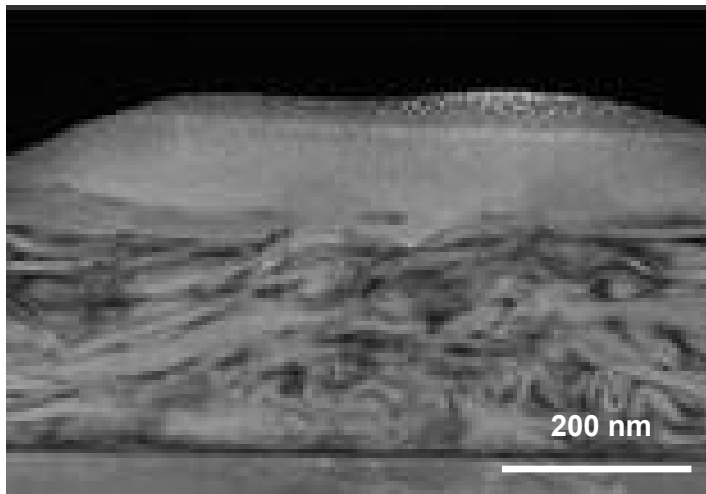
DC Mag pure MoS₂



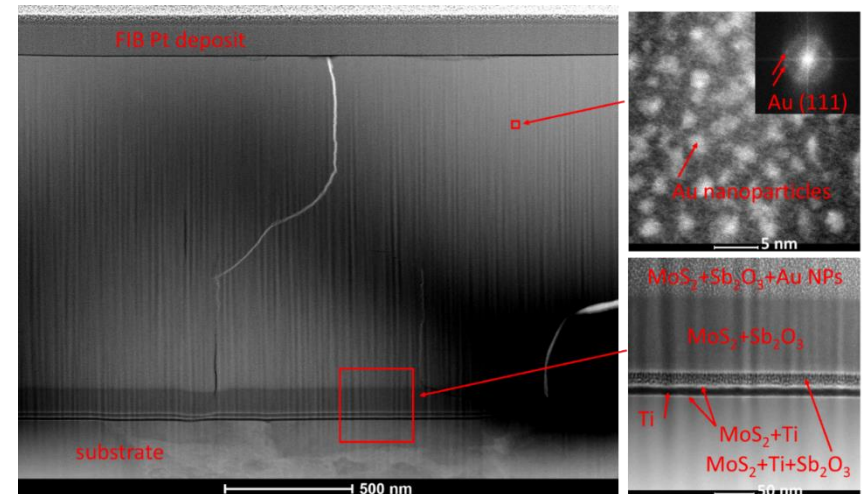
MoS₂/Ti + pure MoS₂



N₂ spray pure MoS₂



MoS₂/Sb₂O₃/Au

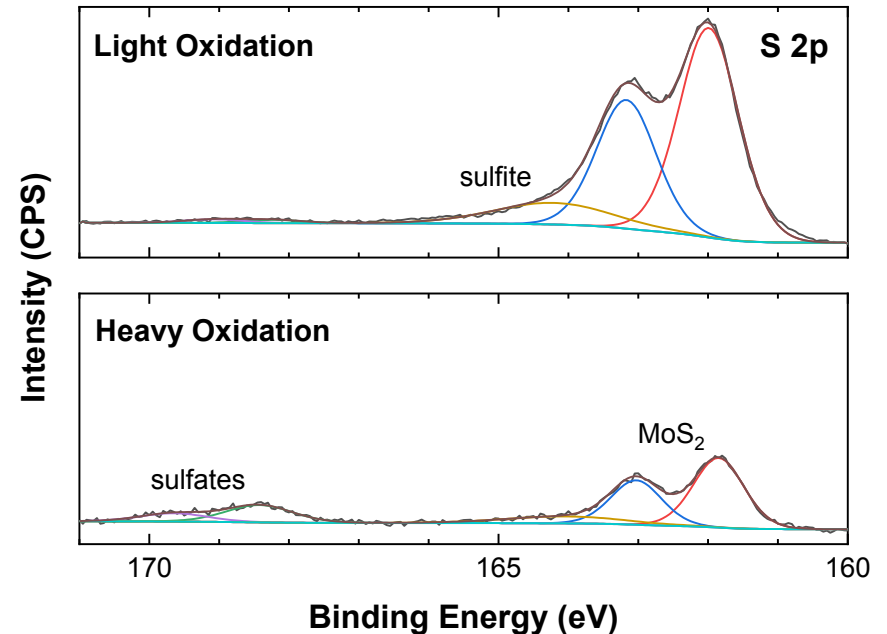
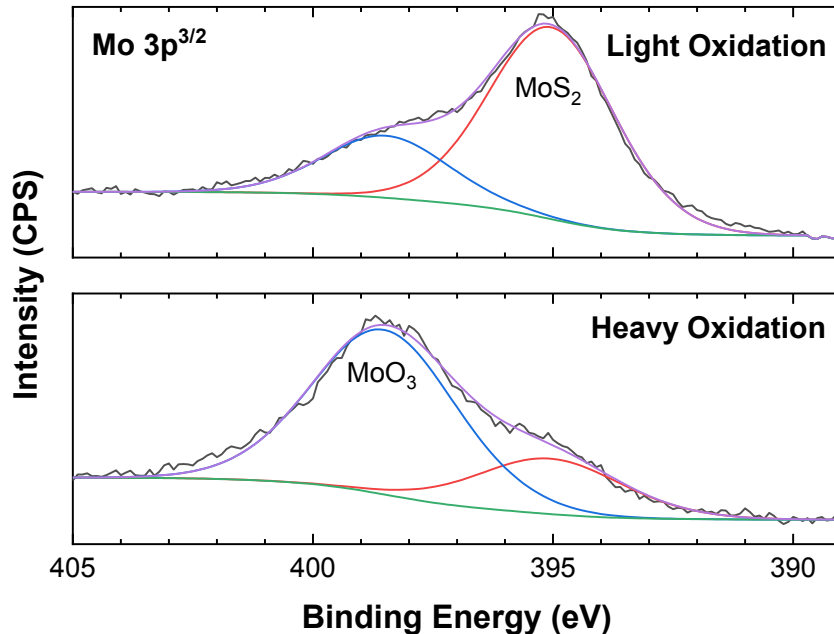


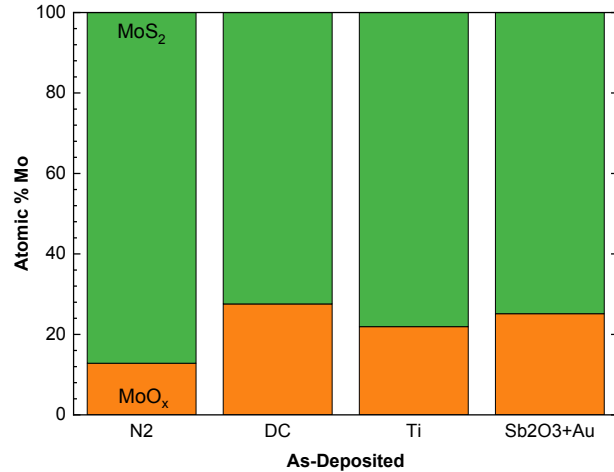
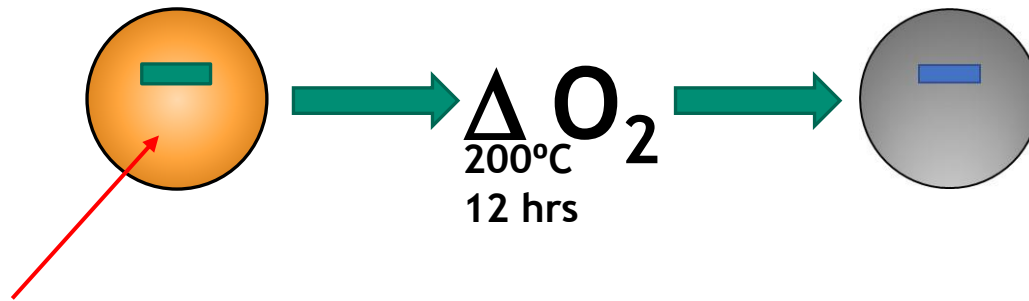
Quantifying MoS₂ Oxidation via XPS



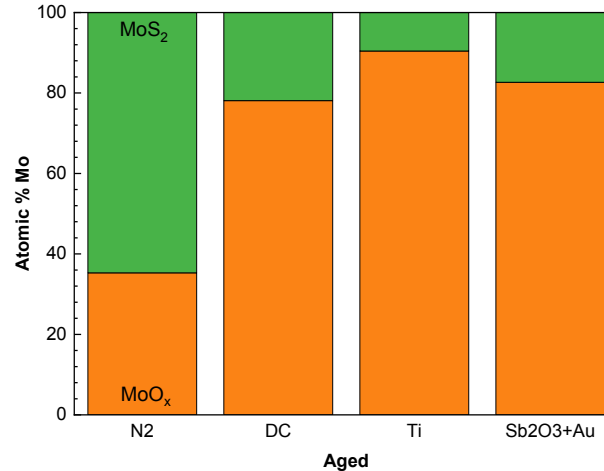
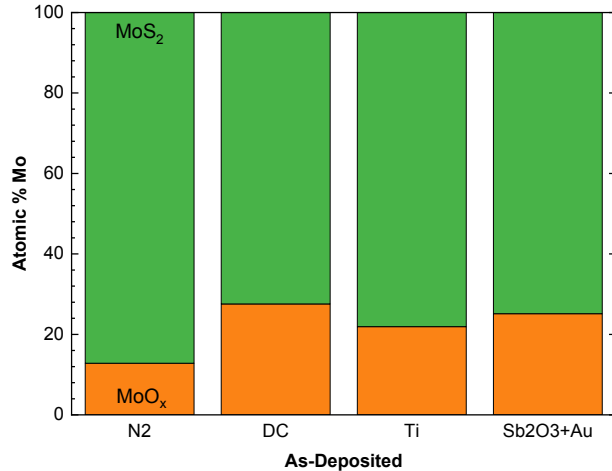
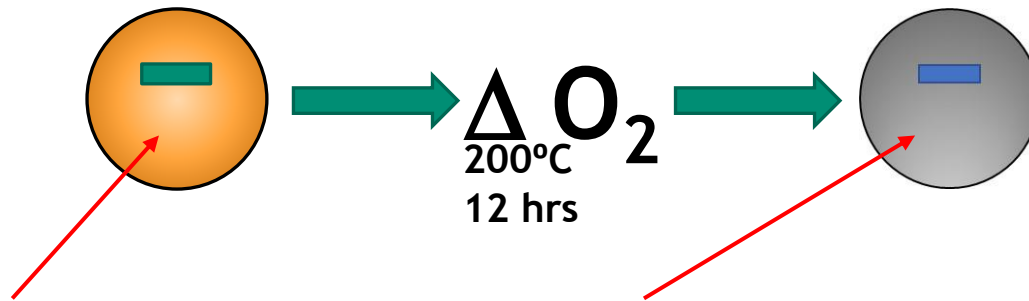
X-ray Photoelectron Spectroscopy (XPS) for surface chemical analysis

- survey scan for concentration of major elements present
- detailed scans of Mo3p, S2p spectral regions
- deconvolution of detailed scans to determine amount of Mo, S bonded to one another compared to oxidized species (MoO₃, sulfates, sulfites, etc.)
- surface sensitive – analyzing the top few nanometers

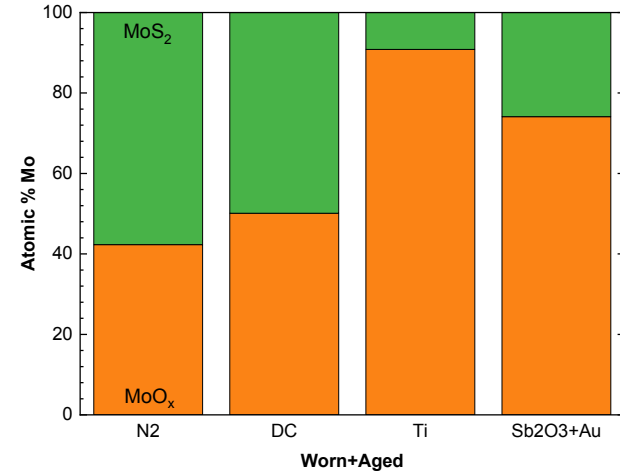
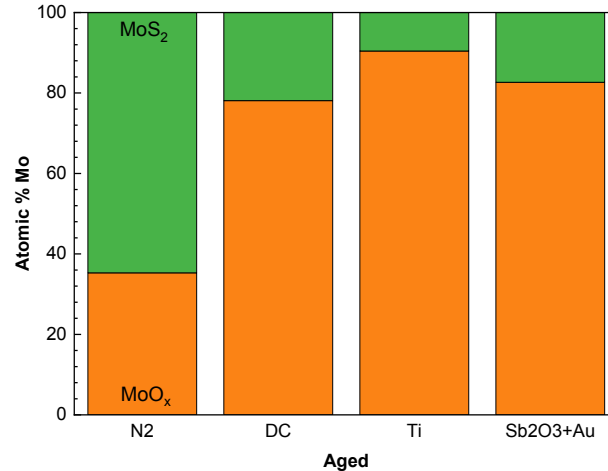
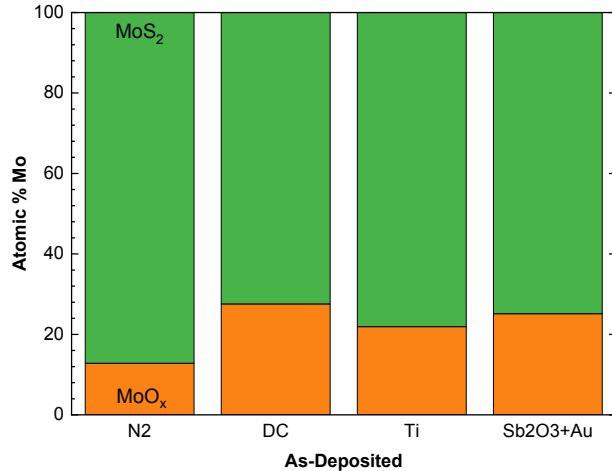
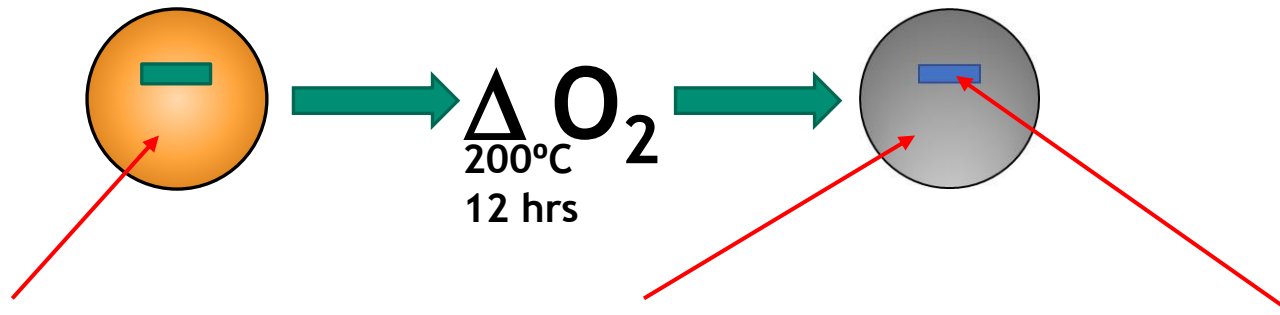




- Sprayed films natively contain less oxide than what is taken up during sputtering

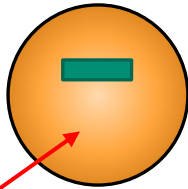


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- Sprayed films in unworn state also less susceptible to oxidation



- Sprayed films natively contain less oxide than what is taken up during sputtering
- Sprayed films in unworn state also less susceptible to oxidation
- Worn area for DC mag films behaves like sprayed films, preventing oxidation
- Composite films run-in do not appear to buy significant protections to oxidation as compared to pure counterparts

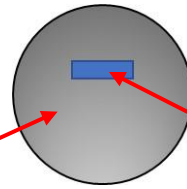
Friction Results – Pure MoS₂ Films



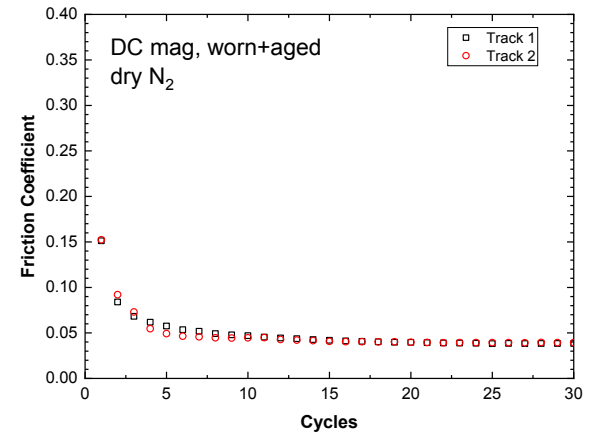
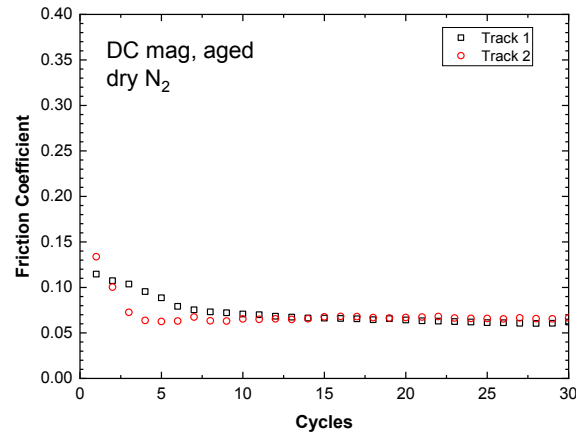
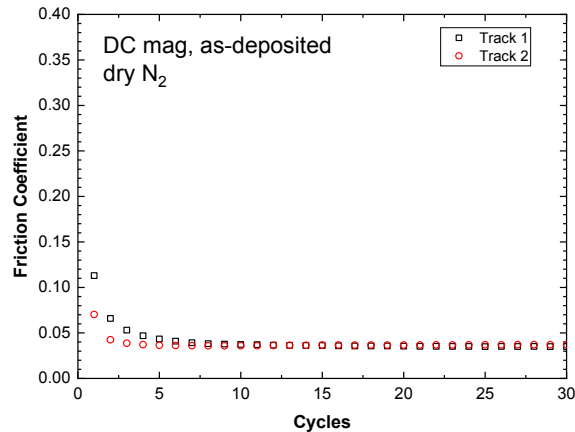
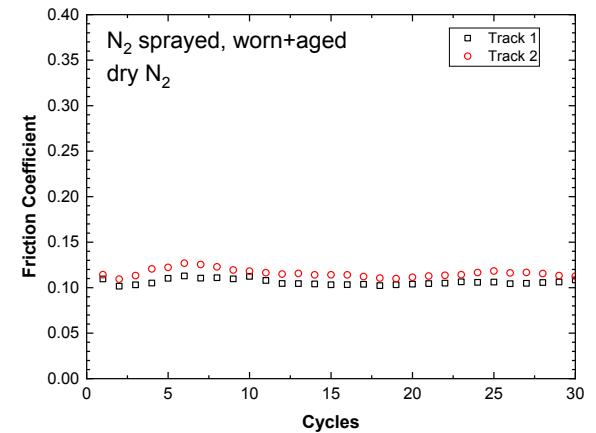
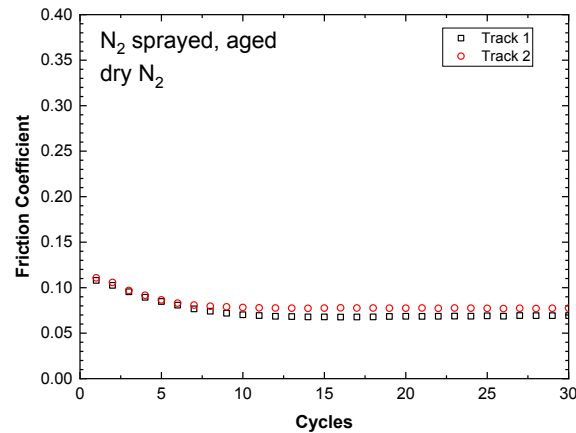
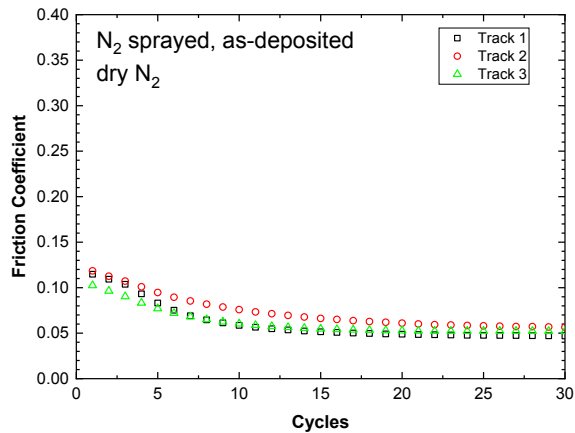
As Deposited



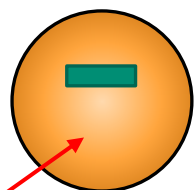
Aged



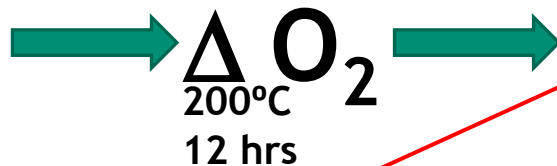
Worn + Aged



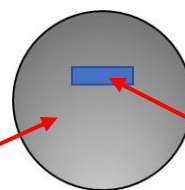
Friction Results – Composite MoS₂ Films



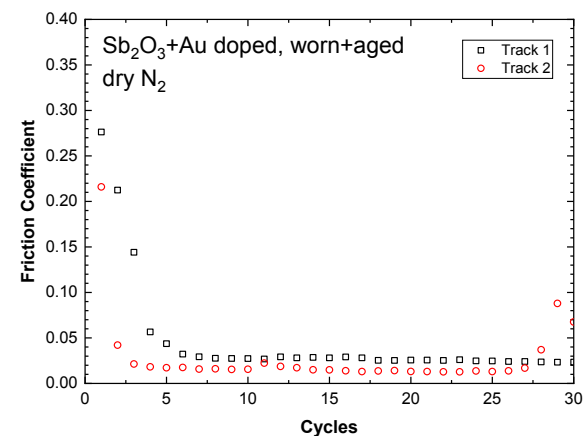
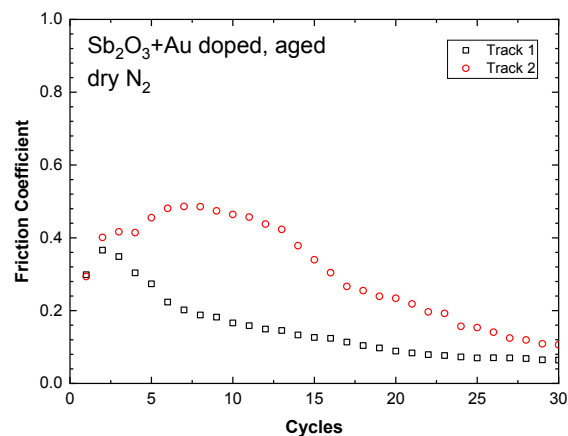
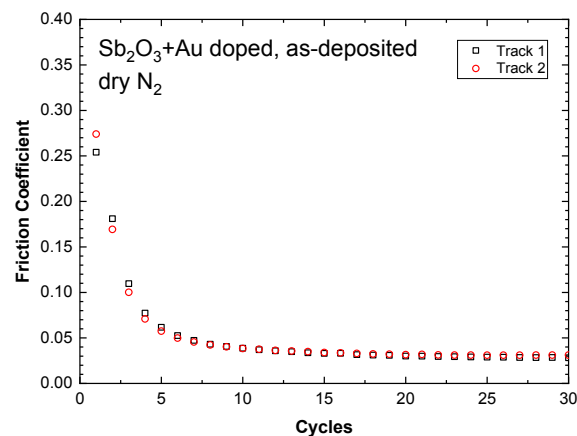
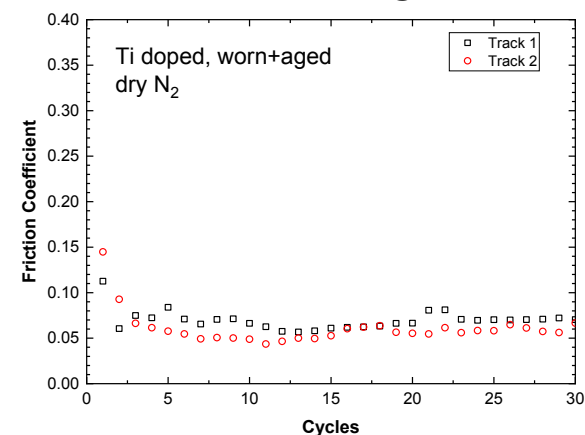
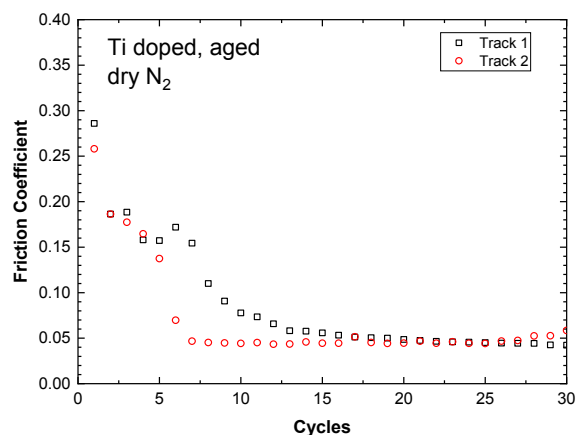
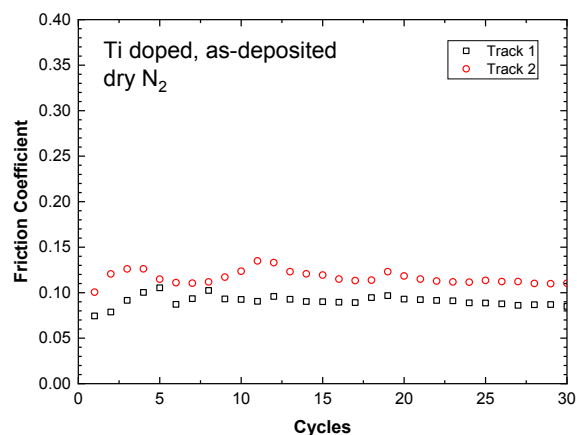
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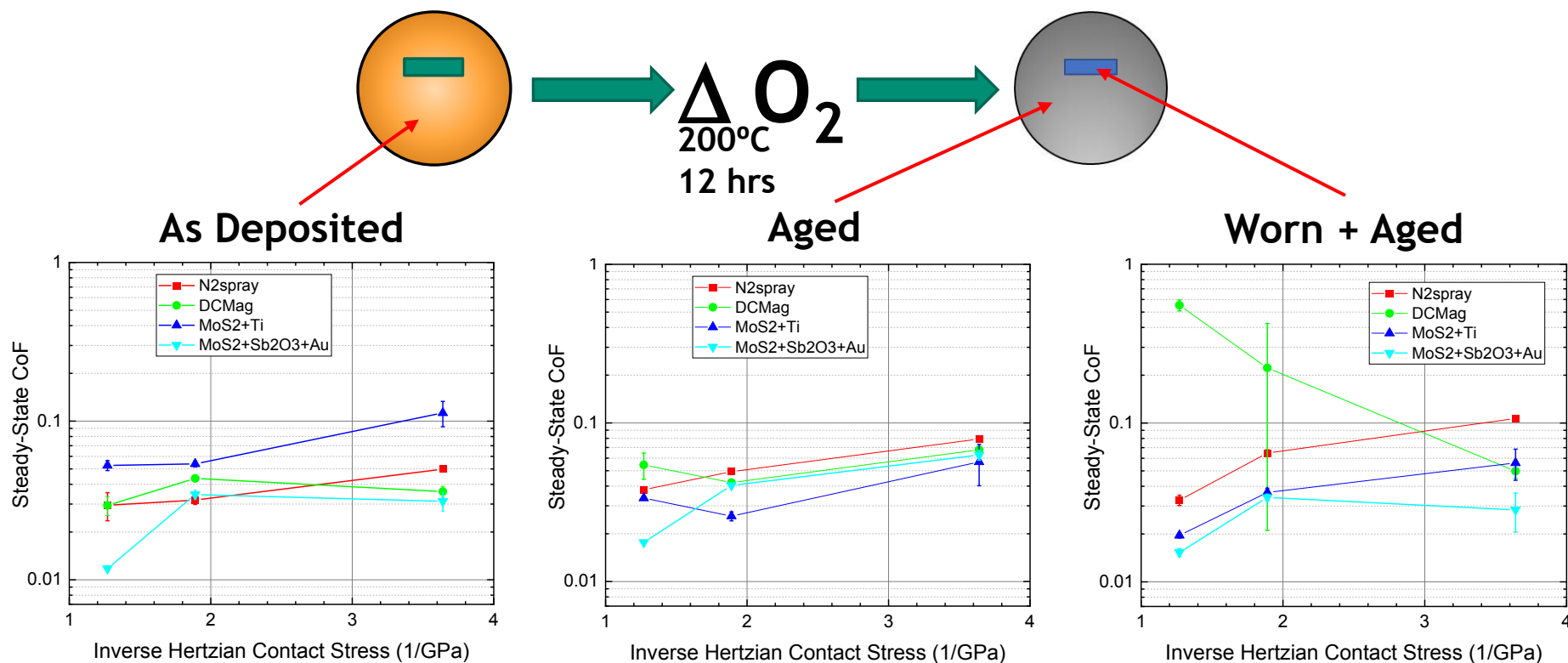


Aged



Worn + Aged



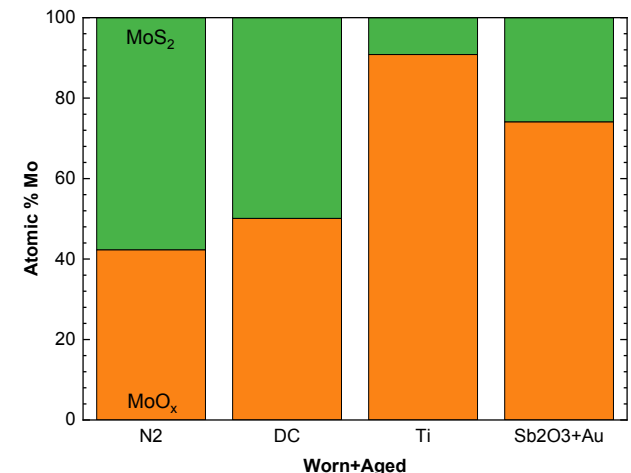
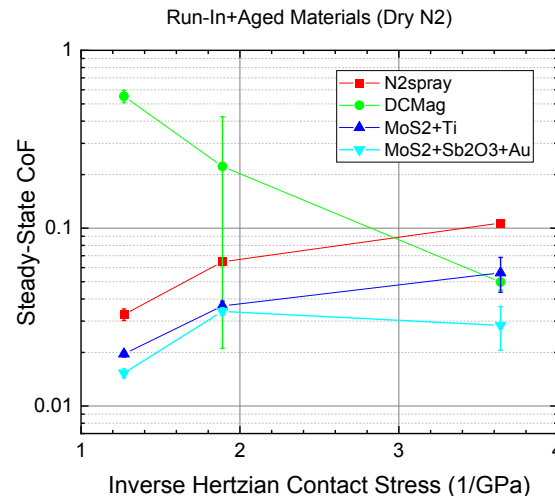
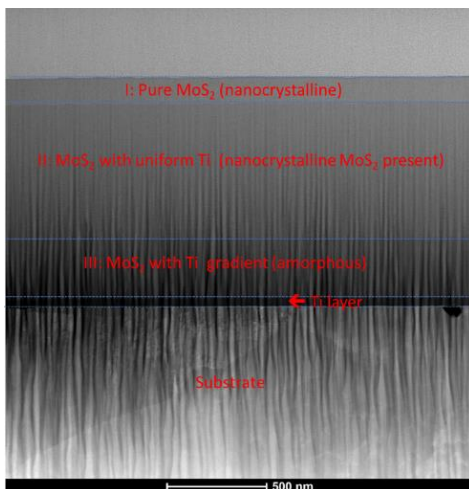


- MoS2-Sb2O3-Au generally lowest friction as deposited and behaved decently after aging
- MoS2-Ti films improved the most after aging
- Sprayed films & DC mag (both pure MoS2) relatively consistent but generally increasing after aging with DC mag exhibiting failures at higher loads

MoS₂ Aging Series Takeaways



- Generally, doped films maintained lower steady state friction coefficients, and improved after aging/run-in prior to aging compared to undoped films
- Undoped films generally exhibit lowest initial friction behaviors compared to doped
- Undoped films also exhibit best oxidation resistance, likely due to lower reactivity of more crystalline materials at film surface (akin to run-in)
- Results suggest it is possible to sputtered MoS₂ films with structure that can resist oxidation & minimize initial friction via surface modification

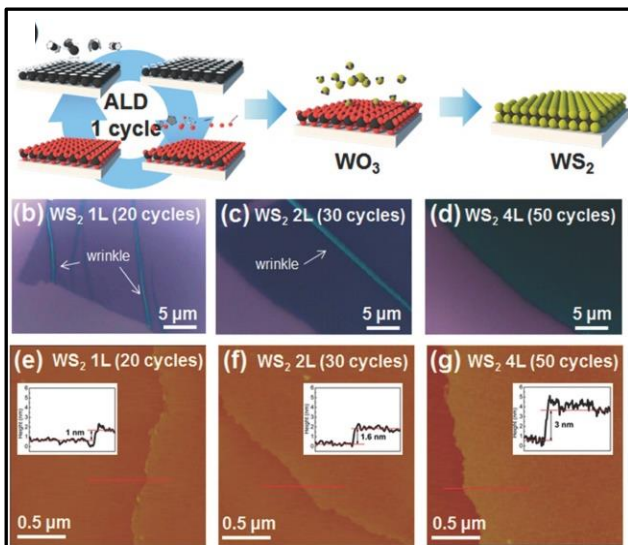
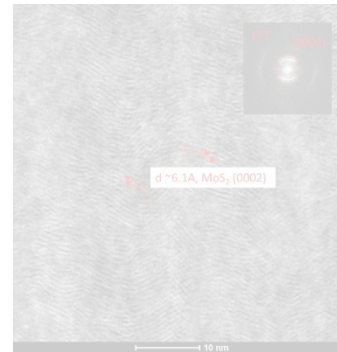
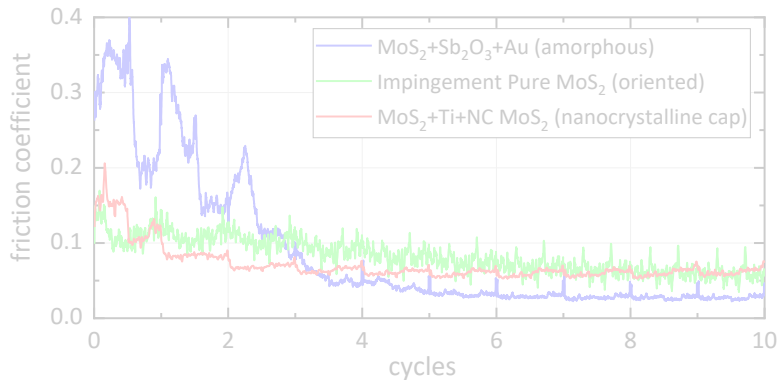


Efforts to Reduce Run-In: Structure/Composition



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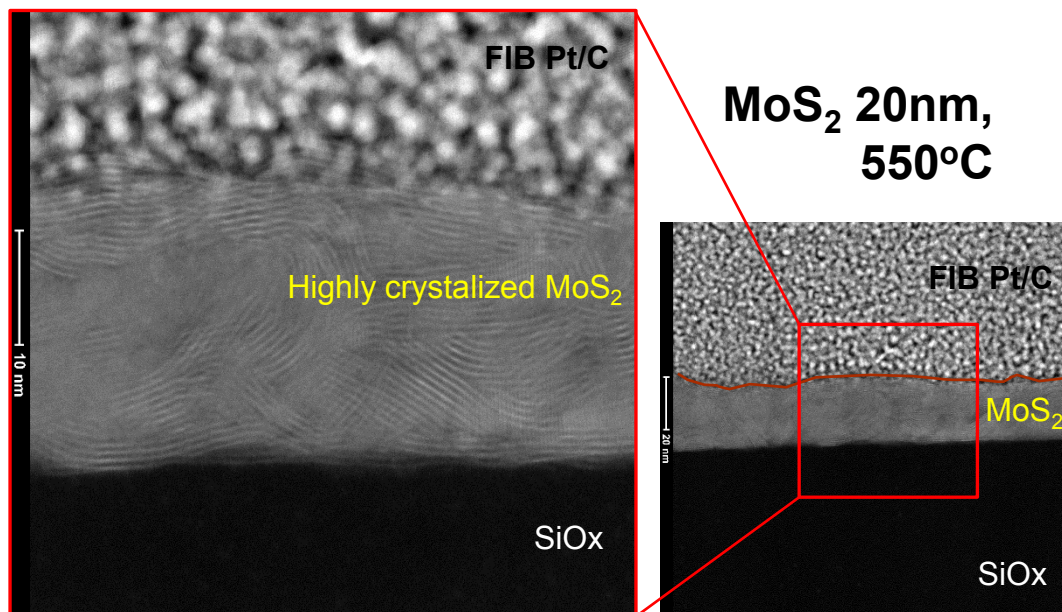
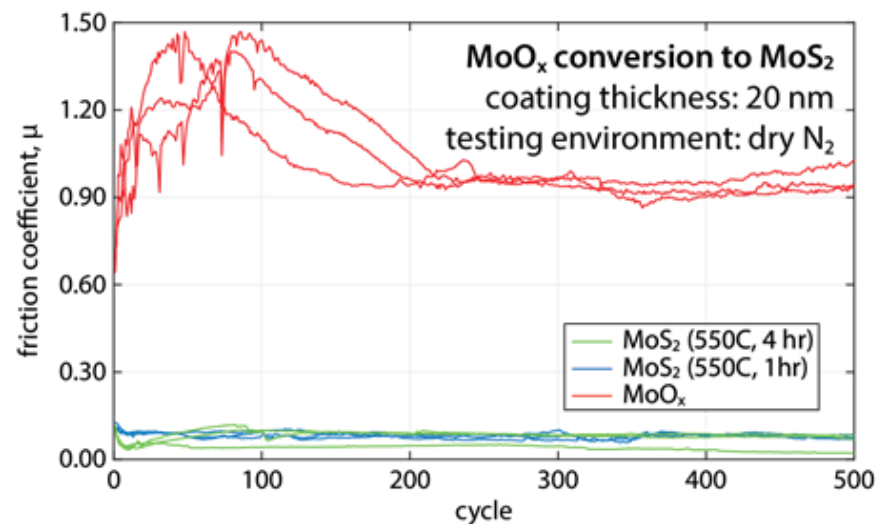
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Deposition/Conversion conditions

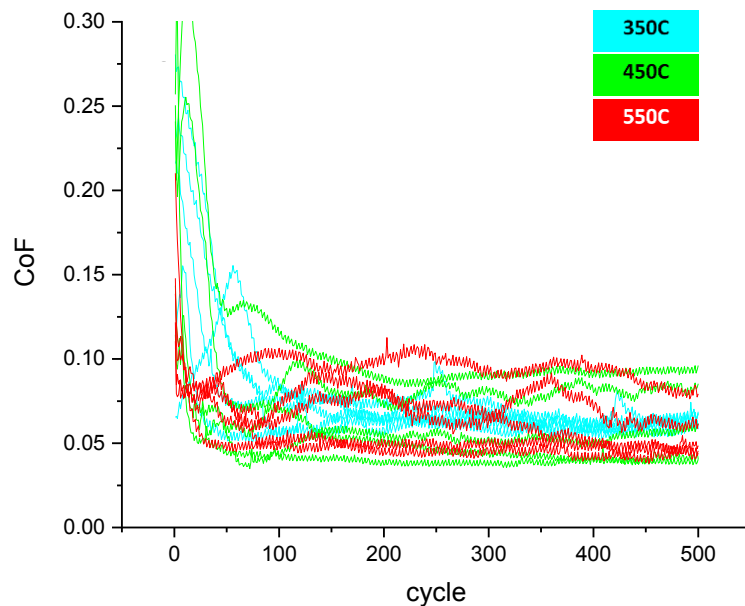
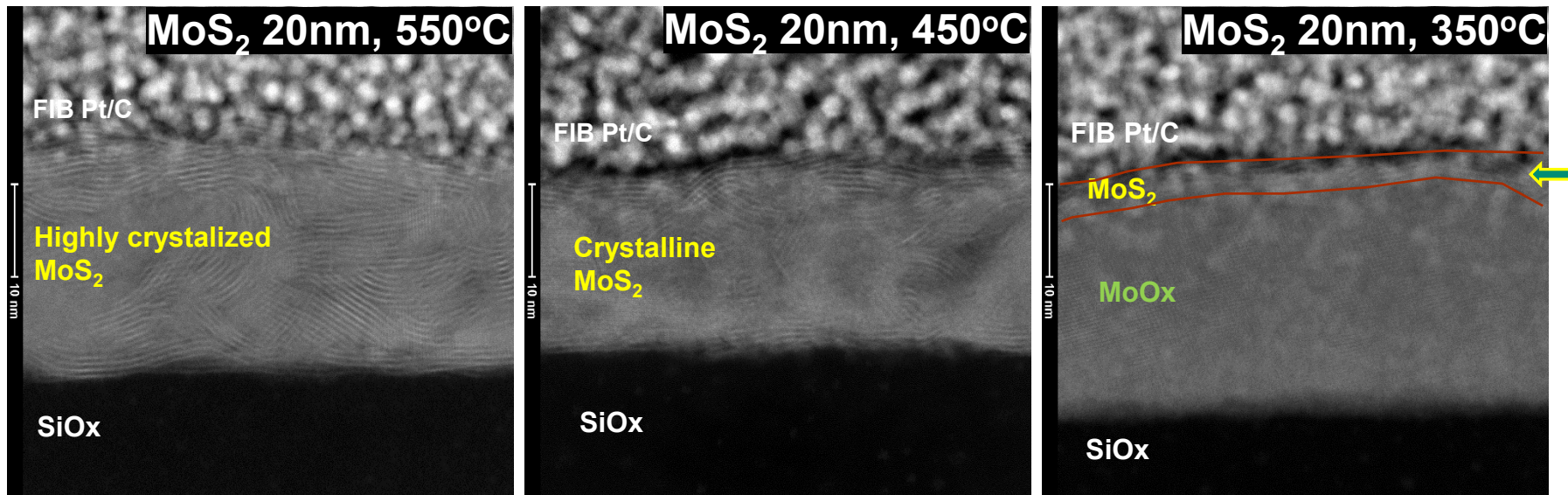
- Moly-hexacarbonyl precursor (AO plasma with water?), ???C for ???hr producing 20nm thick MoO_x on SiO_x wafer
- Initial conversion conditions: 550C for 1-4 hrs, flowing H₂S/Ar mixture gas



TEM/friction observations

- Pre/post friction data shows clear change in friction behavior from oxide to sulfide
- TEM confirm presence of fully converted MoS₂ at 550C conversion temperatures

Conversion at lower temperatures?



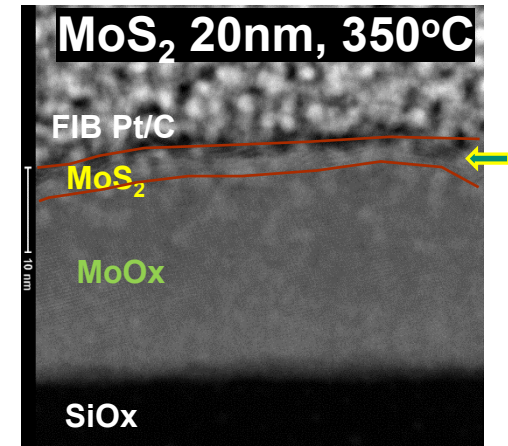
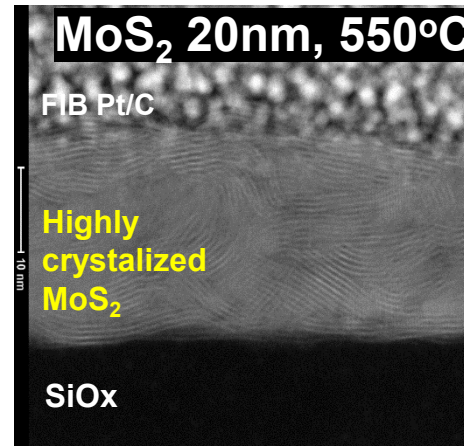
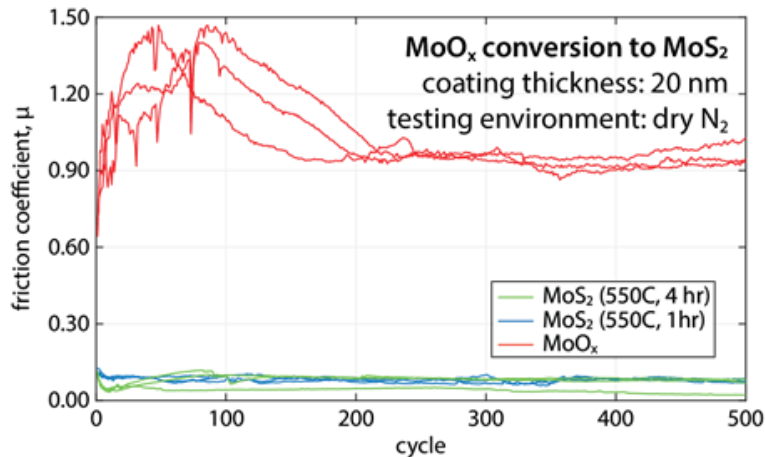
Observations

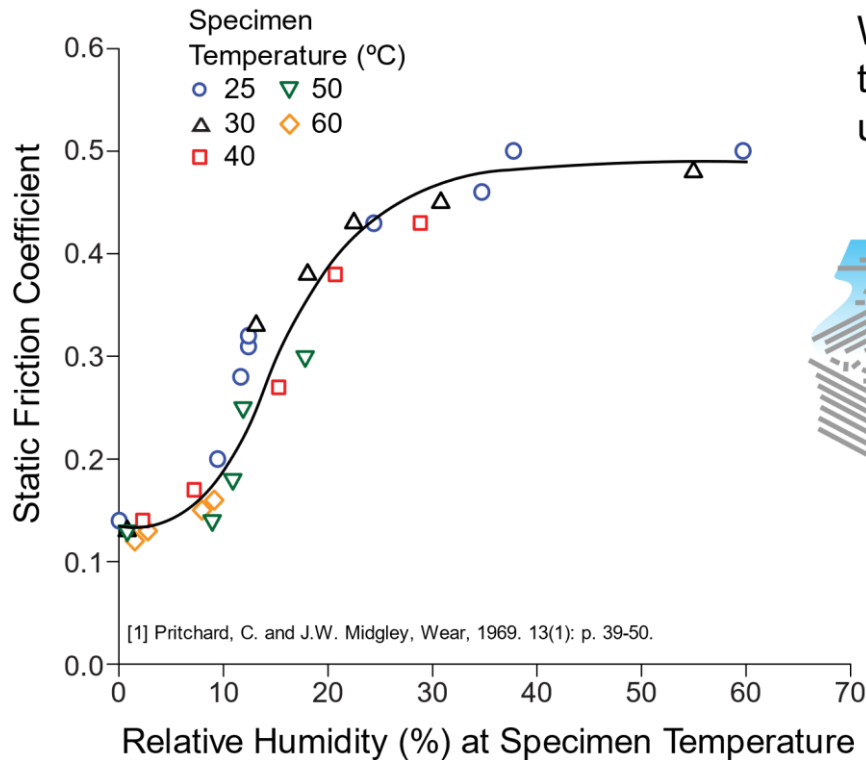
- Friction behavior at all conversion temps indicates presence of MoS₂
- TEM reveals complete conversion at ≥ 450 C
- For 350C conversion, only top few nm have converted to MoS₂ (still providing ~ 500 cycles of low friction)

MoS₂ ALD Takeaways

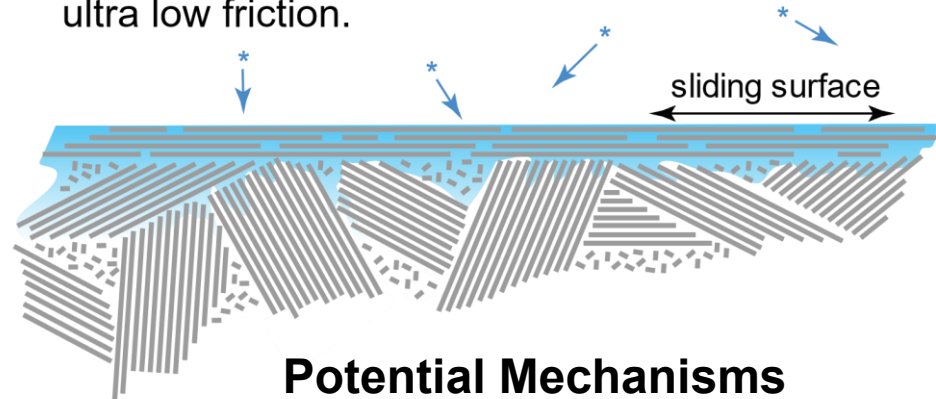


- MoO_x growth and subsequent conversion is a relatively simple ALD process that can produce thin films with excellent friction behavior
- Complete conversion of 20nm oxide observed at 450C and above after 1 hr
- Incomplete conversion at 350C suggests kinetics not adequate, yet thin layer of MoS₂ still provides lubrication
- Additional work required to understand if conversion is possible at lower temperatures to enable deposition on work parts (i.e. steels)





Water adsorption and diffusion throughout a surface layer interrupts ultra low friction.



Potential Mechanisms

- Adsorption (polar bonding, capillary forces, edge interactions, etc...)
- Oxidation (H_2O vs O_2 , high temp, etc...)

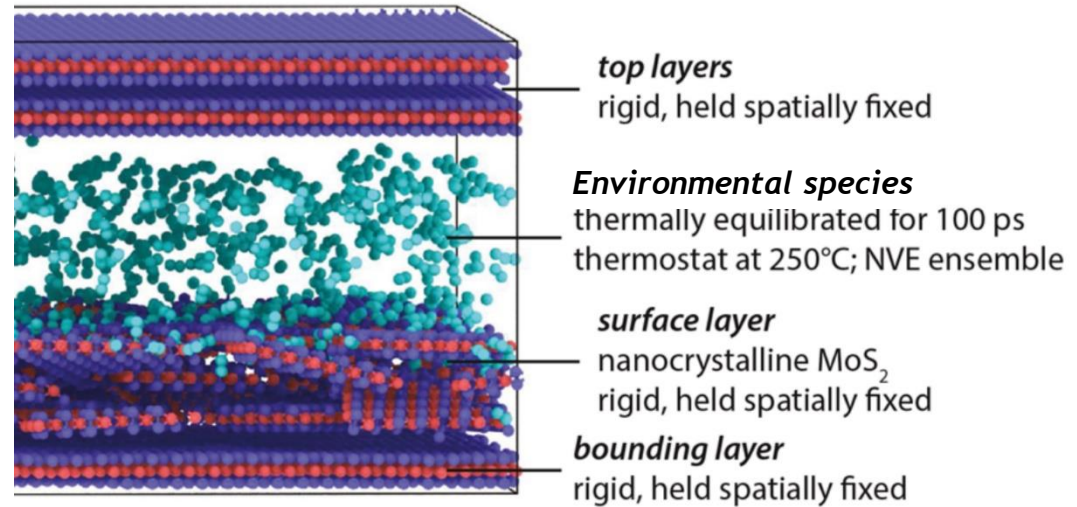
Little is known about how water fundamentally interacts and influences friction behavior in MoS₂

Structurally Driven Environmental Degradation of MoS₂

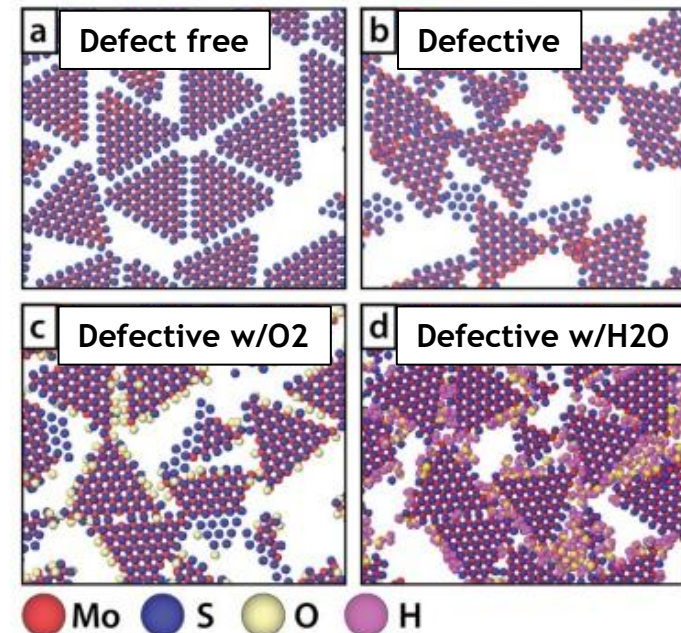
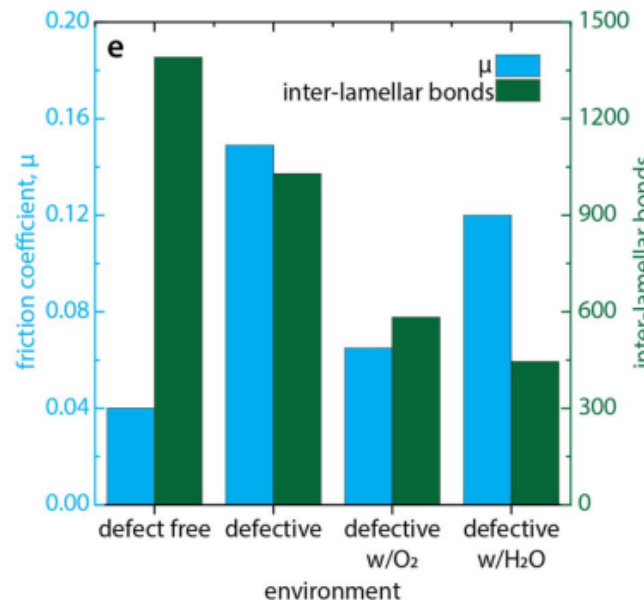


MD simulations show us a few things:

1. Sliding on pristine MoS₂ orders lamella and increases lamella size – low friction
2. O₂ and H₂O passivate edge sites preventing coalescence lamella



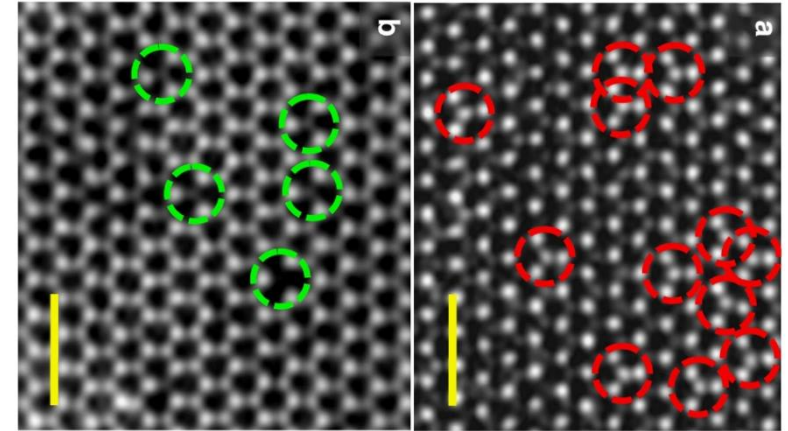
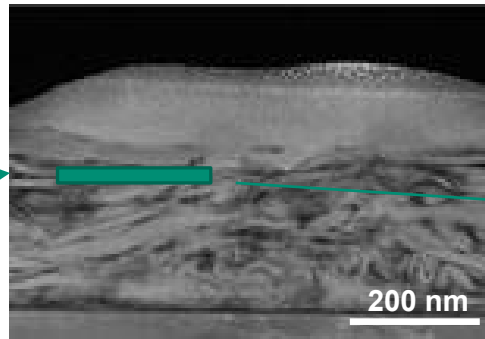
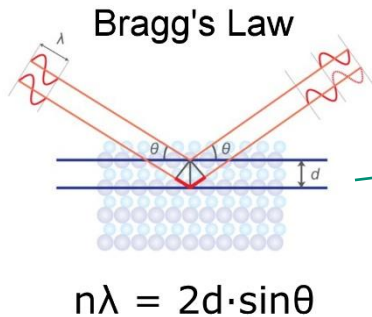
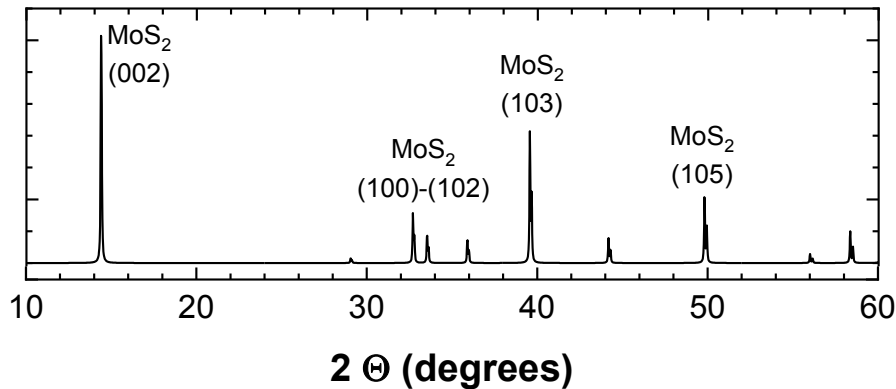
Takeaway
MD suggests
changes to surface
structure through
environmental
interactions dictate
friction



Diagnosing Surface Structure is Difficult



X-Ray Diffraction



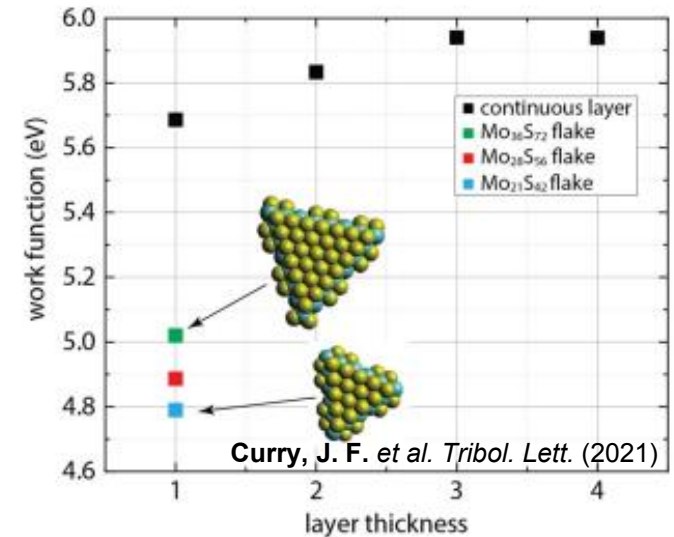
STEM-ADF (Hong et al. Nat. Comm. 2015)

XRD can easily prove crystallite thickness, not width and depth

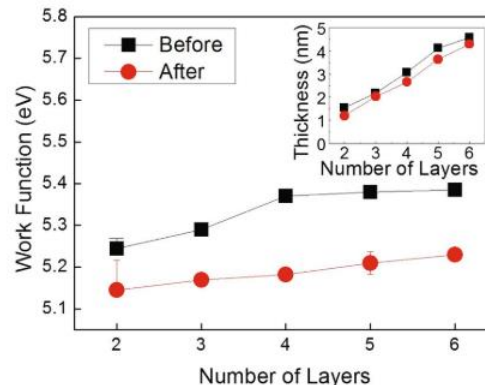
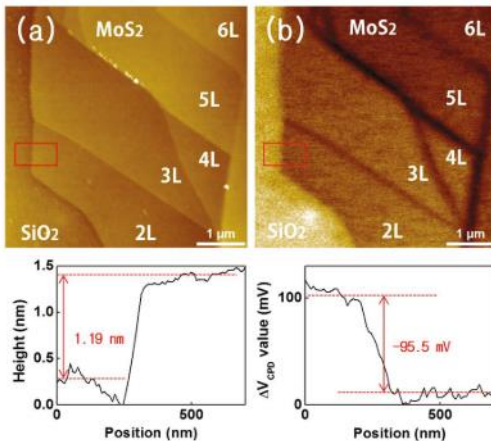
Existing methods for characterizing surface structure and defects (i.e. XRD, TEM, STM, etc) still make it difficult to assess near surface crystallite distribution and defect density.

Using work function to observe changes in microstructure

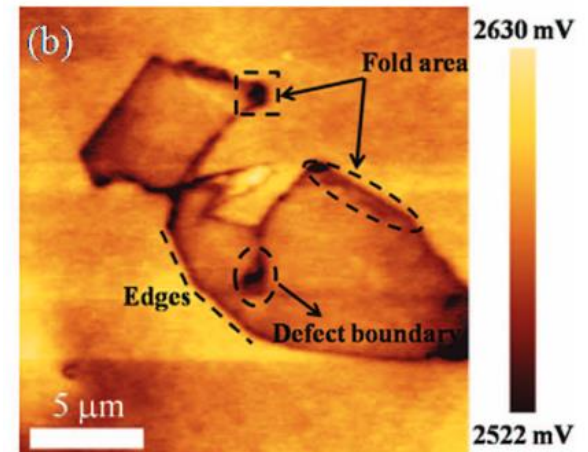
- Work function is a property of the surface
- Can be measured from KPFM, PEEM in combination with UPS
- Scales with number of layers
- Scales with the size of MoS₂ lamella



Role of thickness & adsorbates



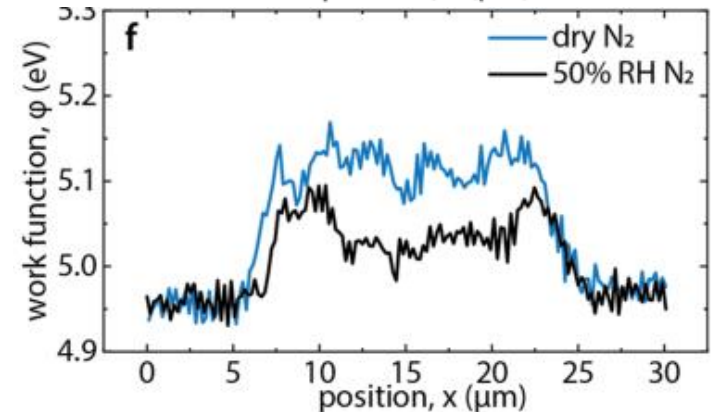
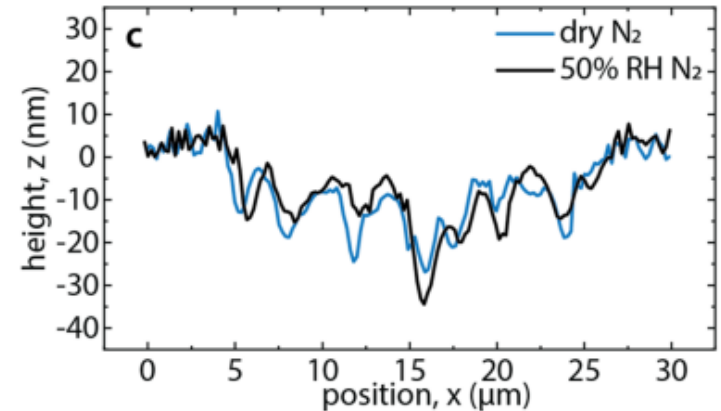
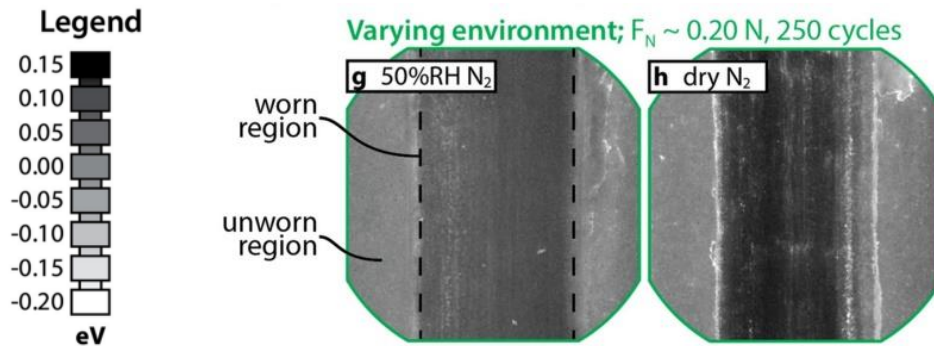
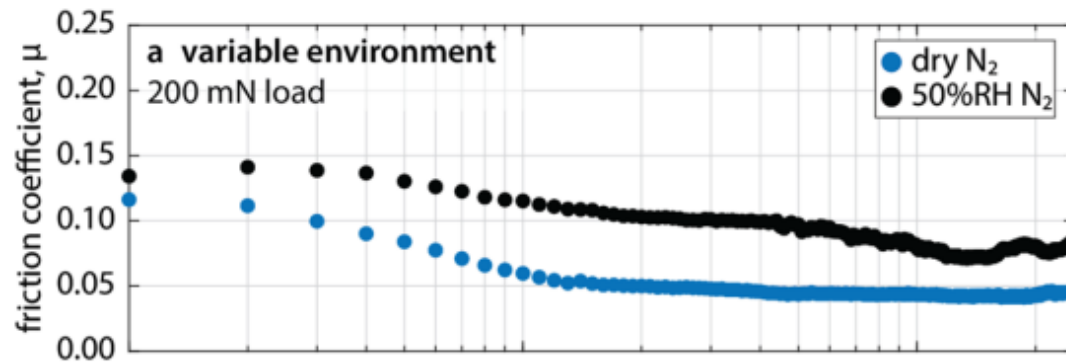
Role of grain boundaries & edges



Hao et al. AIP Adv. (2013)

Takeaway: Work-Function can be used to assess changes in surface microstructure

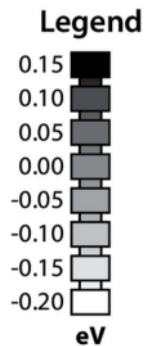
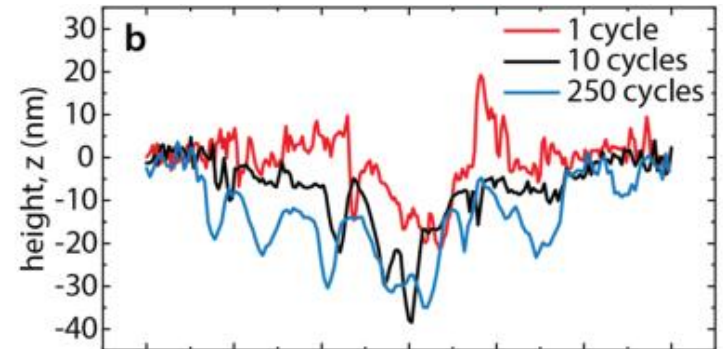
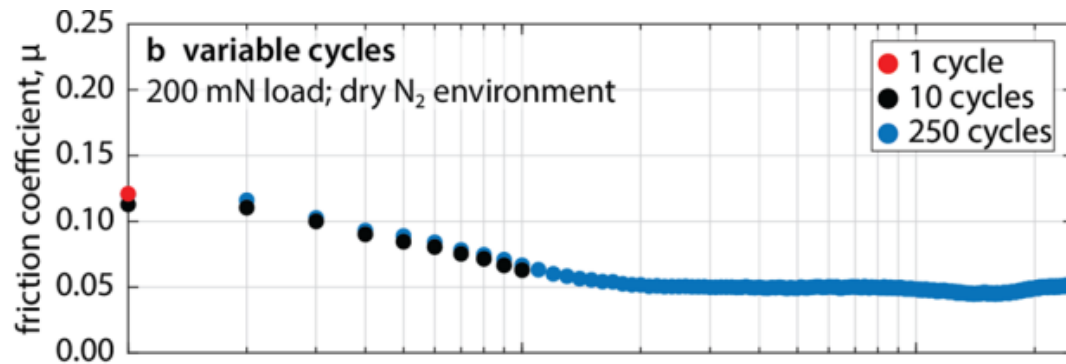
Friction response driven by changes in structure



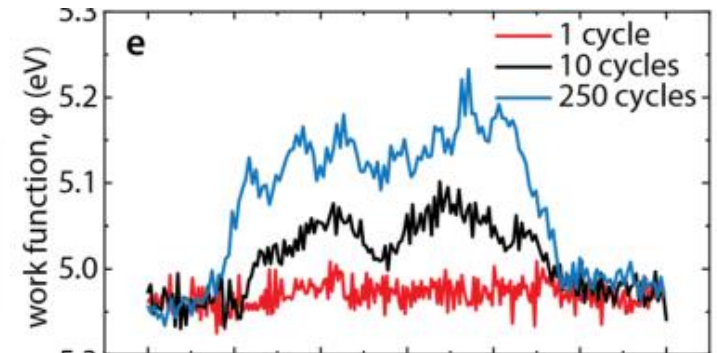
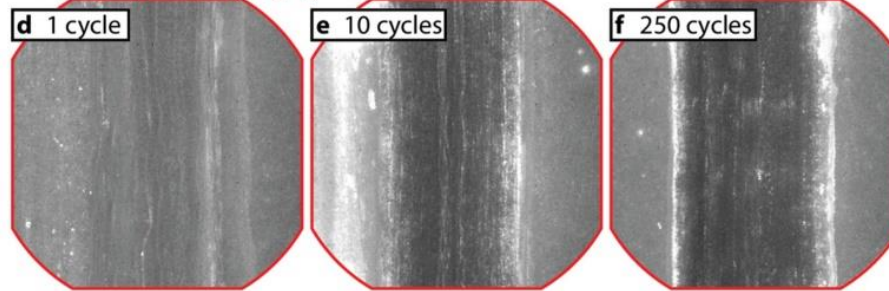
KPFM and PEEM indicate that shear in presence of H_2O still increases work function compared to bulk, but less than sliding without H_2O

Takeaway: H_2O decreases work function by inhibiting formation of large defect free lamella

Friction response driven by changes in structure

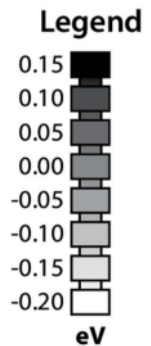
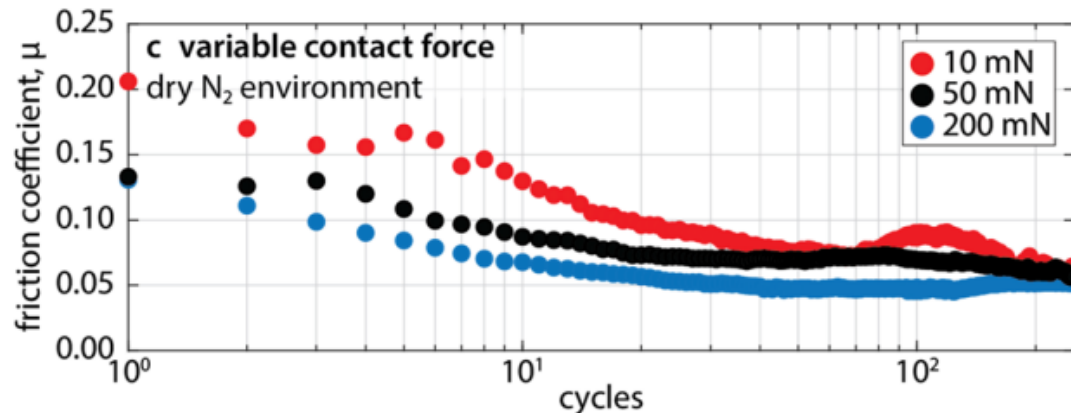


Varying cycles; dry N₂, F_N ~ 0.20 N

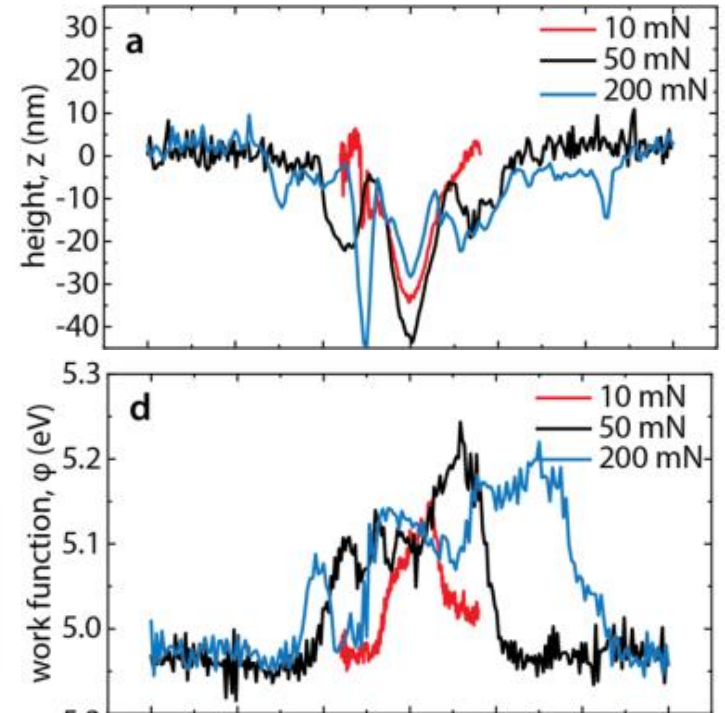
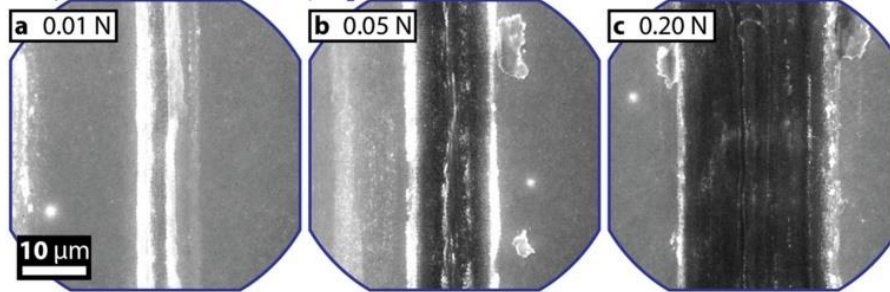


Takeaway #2: The transition to low friction “run-in” is a result of shear combining and reorienting lamella

Friction response driven by changes in structure



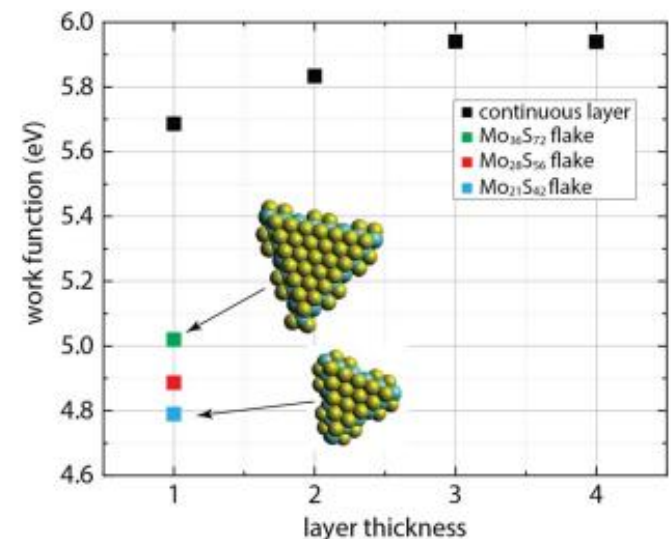
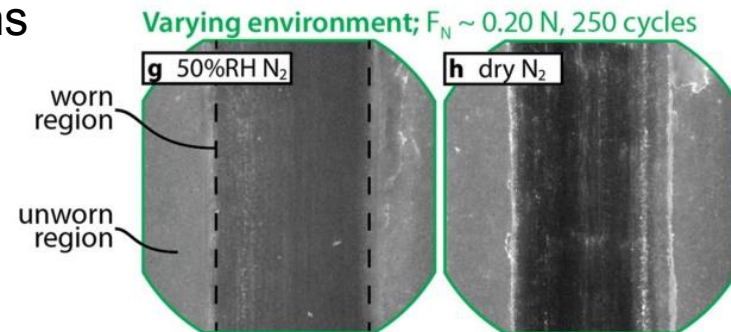
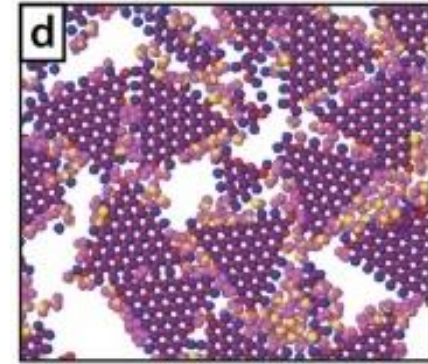
Varying normal load; dry N₂, 250 cycles



Takeaway #1: Increasing contact pressure forms larger lamella

Work Function on MoS2 Take-Aways

- MD simulations suggest water interaction/agglomeration with edge sites prevents formation of long range order MoS2
- Structural degradation (smaller flake size, higher defect density) due to environmental interactions leads to increased friction
- Work function can be used to probe changes in structure at the surface at macroscale
- DFT and literature show that higher work functions are related to larger crystallites and higher layer counts (thickness) of MoS2
- Lower work functions were observed in wear scars associated with higher friction conditions (humidity, low load/cycle count) likely due to less ordered structures

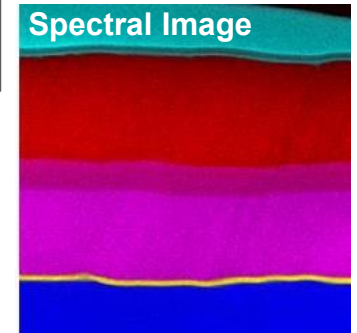
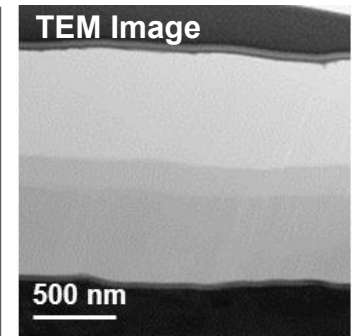
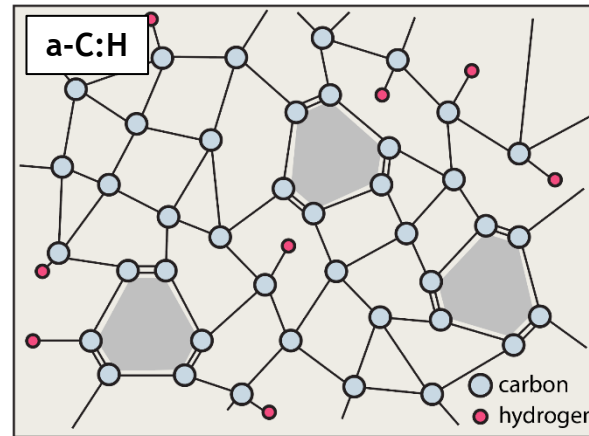
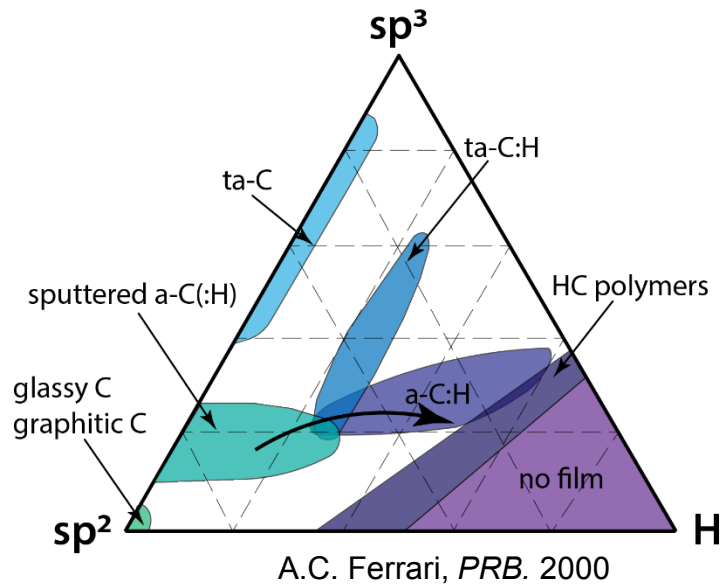




Understanding Variability in Run-in for DLC Coatings

A decorative horizontal bar with a series of small, multi-colored squares (blue, yellow, green, pink, grey, purple, orange, red, blue, yellow, pink, grey) is positioned below the title.

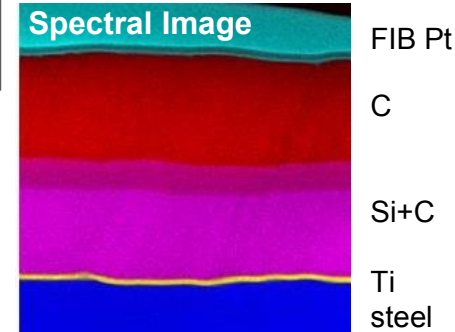
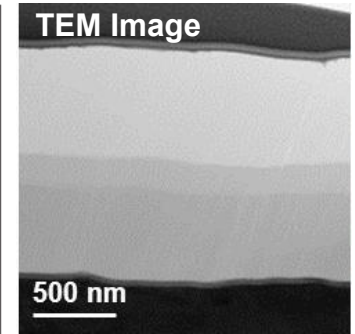
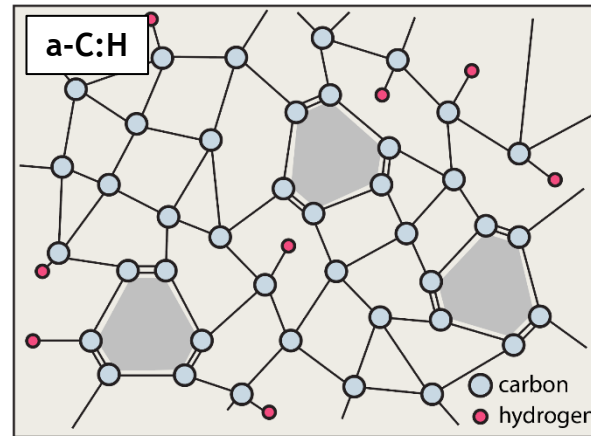
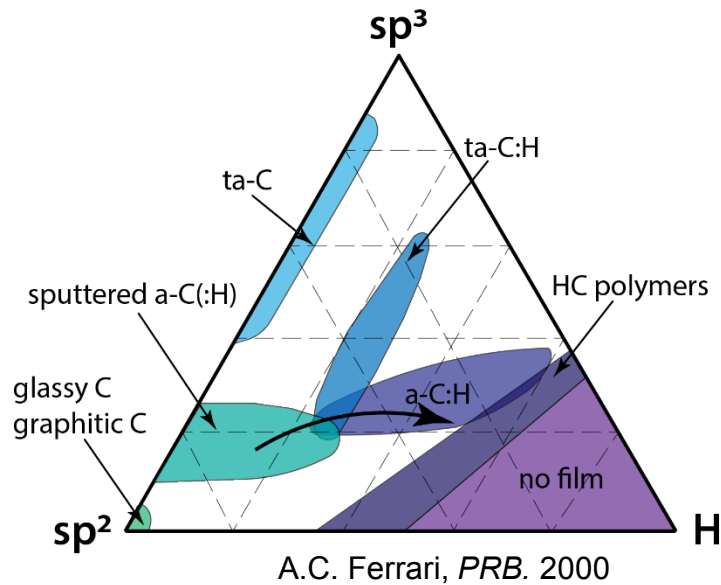
Diamond-Like Carbon (DLC) Coatings



FIB Pt
C
Si+C
Ti
steel

- Amorphous network of $sp^2/sp^3/H$ or other dopant

Diamond-Like Carbon (DLC) Coatings

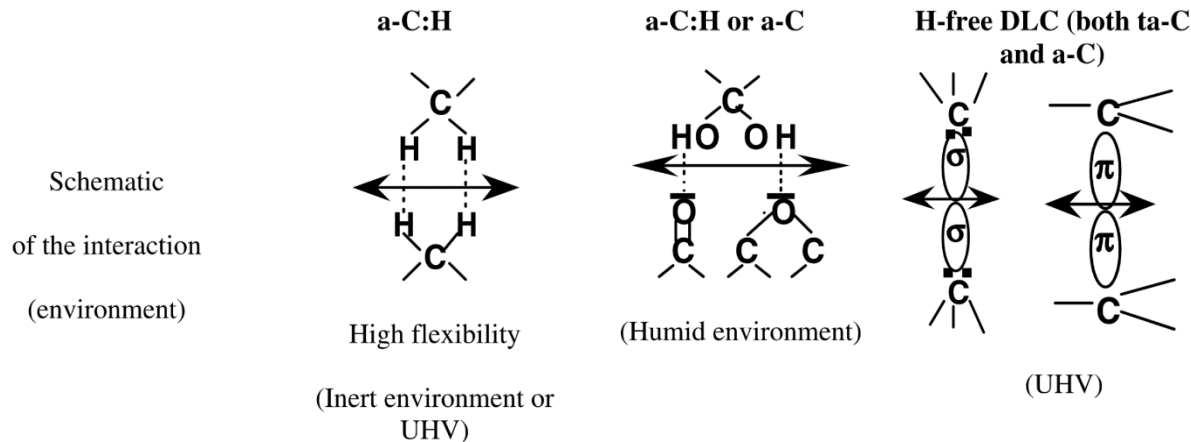


Images courtesy of Mike York, NSC

- Amorphous network of sp²/sp³/H or other dopant
- PECVD process decomposes precursor hydrocarbon gases to deposit carbon films
- Result in complex film compositions with varying tribological behavior...

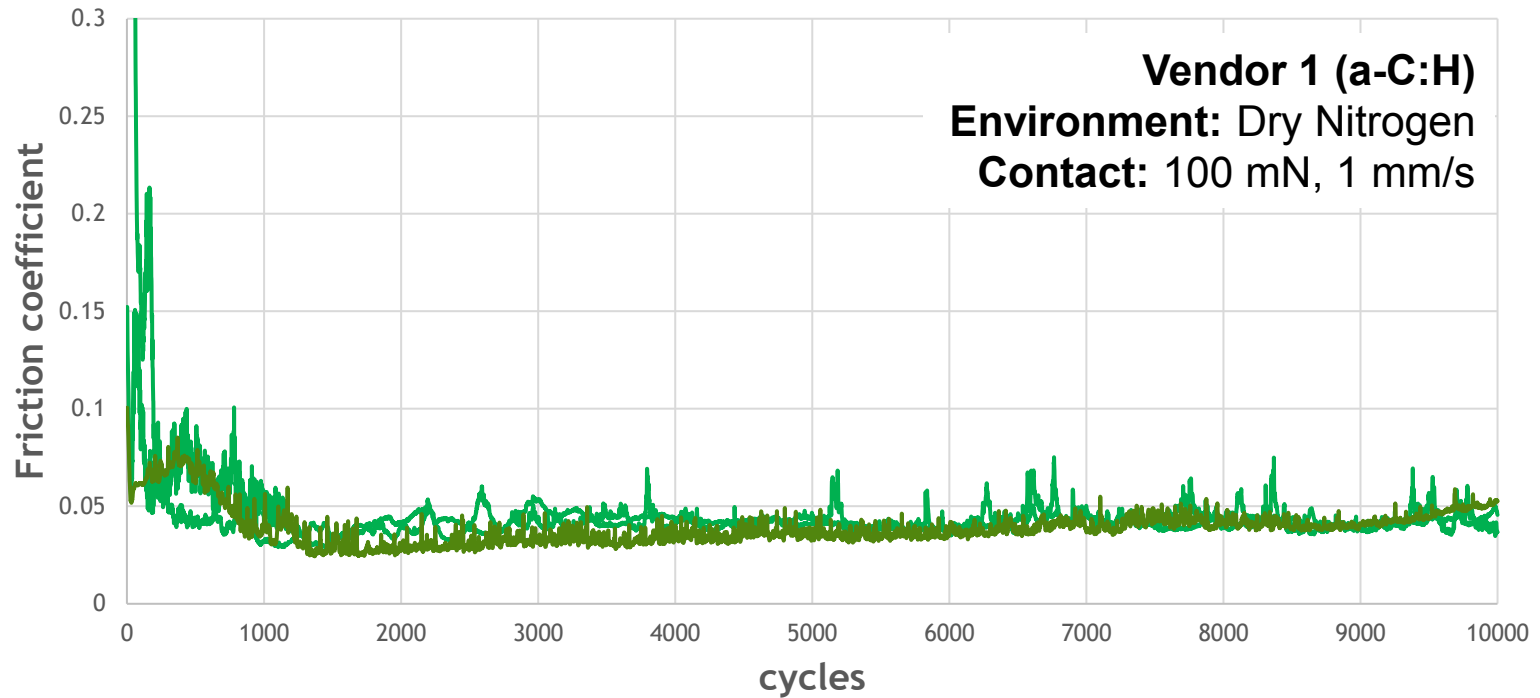


solid lubricant	deposition methods	μ_{ss}	F_n	environment
graphite (sp^2 bonding)	evaporation, pyrolysis of HC polymers	0.2 - 0.5	0.5 N - 1 N	dry N_2 /UHV
		0.1 - 0.2	0.5 N - 1 N	humid air
DLC (mixed sp^2/sp^3 bonding)	rf and dc sputtering, ion beam, CVD	0.6 - 0.7 a-C	10 N	dry N_2 /UHV
		0.001 - 0.05 a-C:H	10 N	dry N_2 /UHV
		0.1 - 0.2 a-C	10 N	humid air
		0.2 - 0.3 a-C:H	10 N	humid air

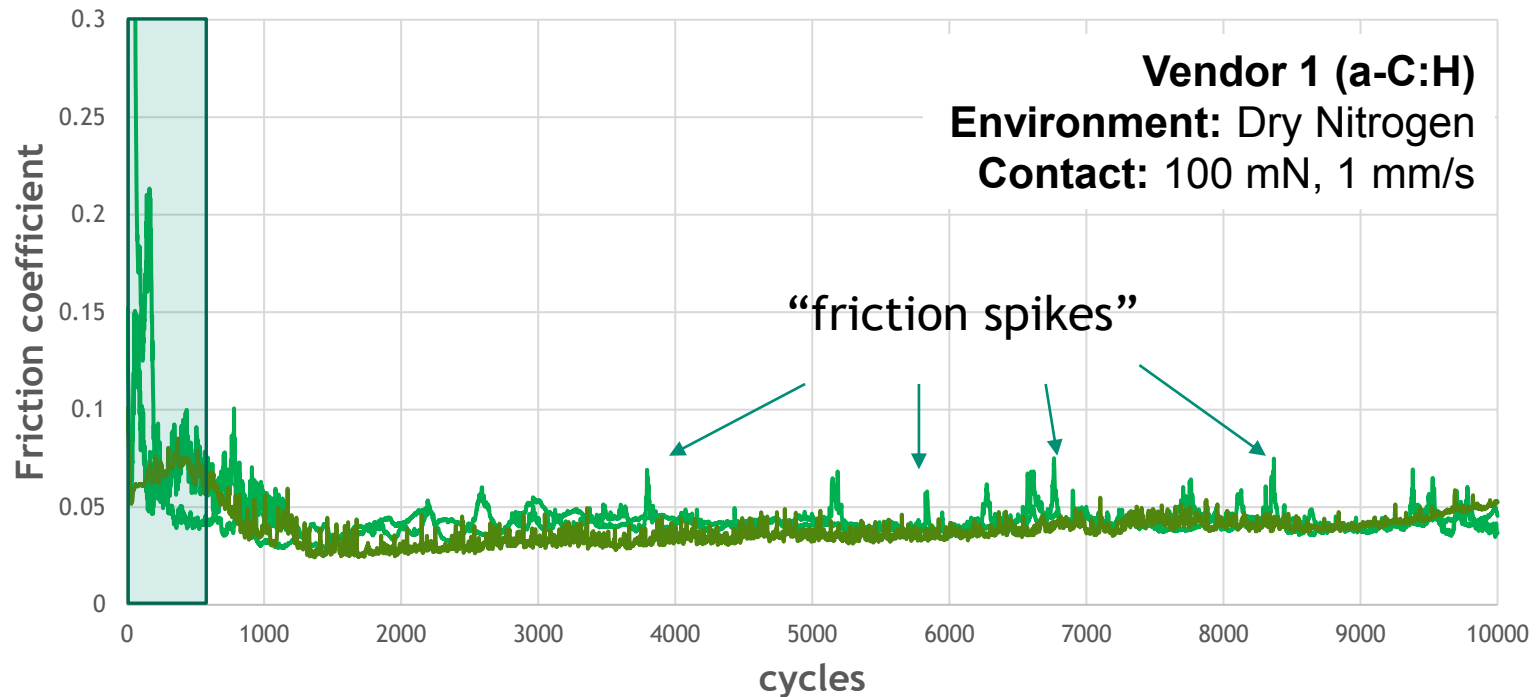


(Scharf, 2013, J. Matl. Sci.)
 (Robertson, 2002, Matl. Sci. & Eng.)
 (Erdemir, 2006, J. Phys. D: Appl. Phys.)

Typical DLC Friction Trace

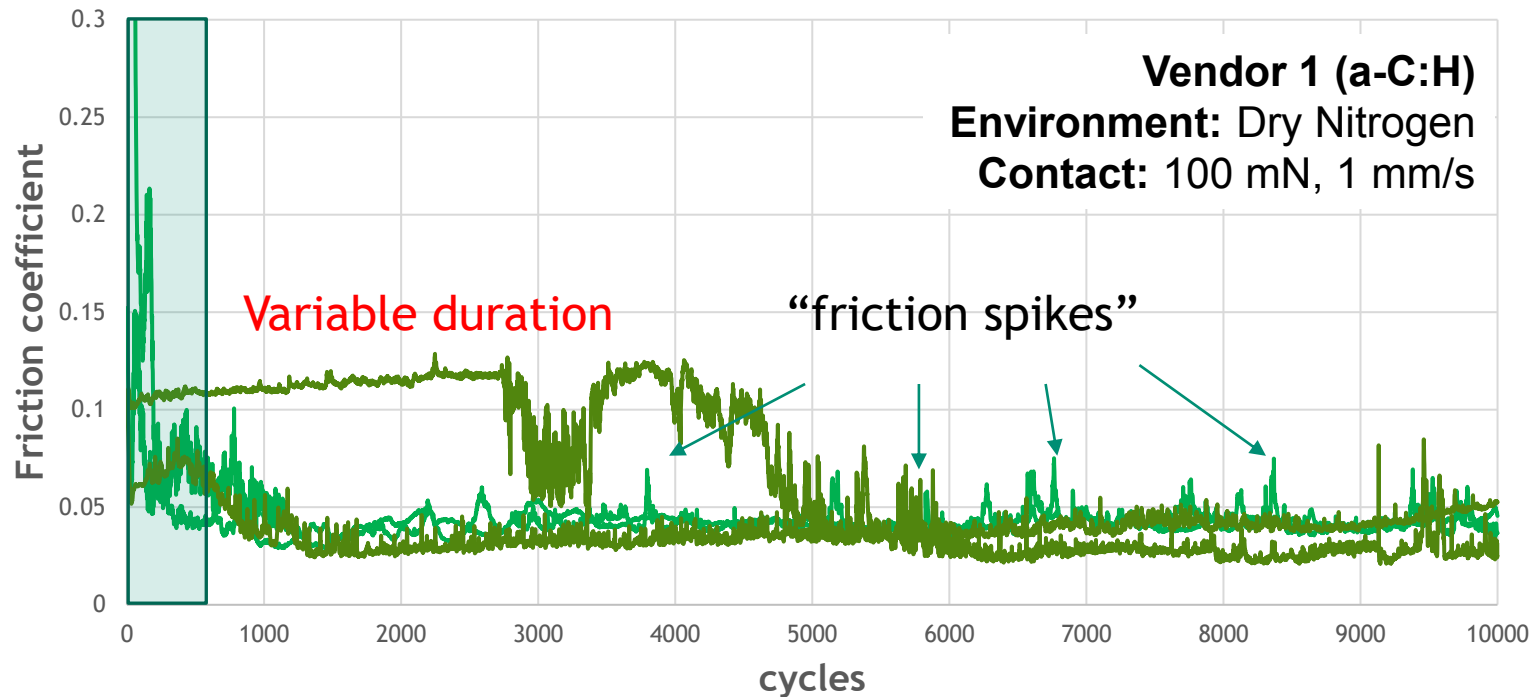


Typical DLC Friction Trace



- Coatings exhibit low steady state friction and occasionally “spikey” steady state behavior
- Magnitude of initial friction may vary between $\mu=0.10$ - 0.30 ...

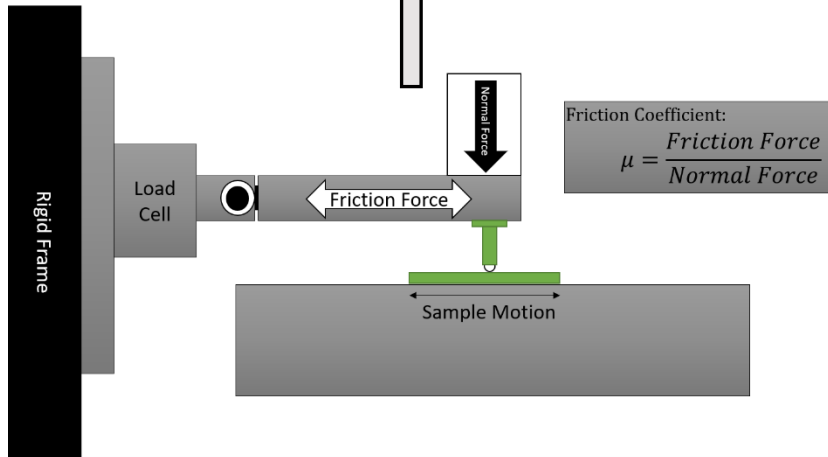
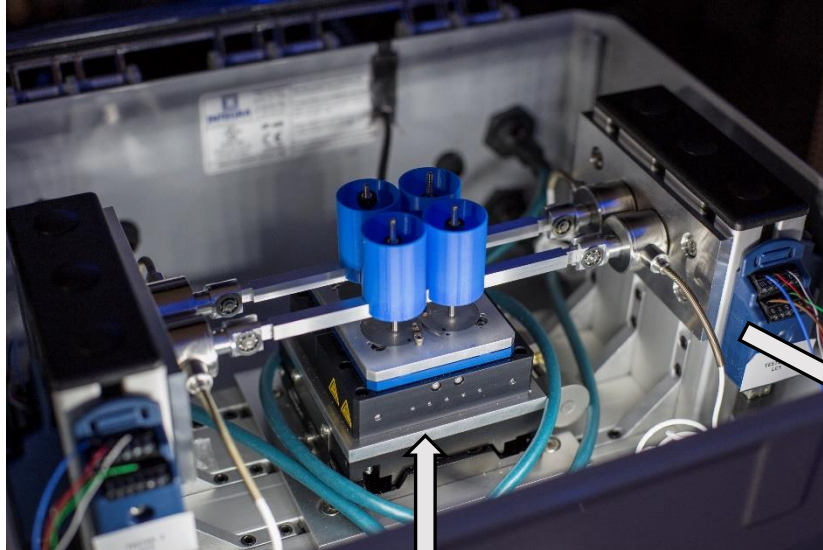
Typical DLC Friction Trace



- Coatings exhibit low steady state friction and occasionally “spikey” steady state behavior
- Magnitude of initial friction may vary between $\mu=0.10-0.30$... run-in duration varies extensively as well

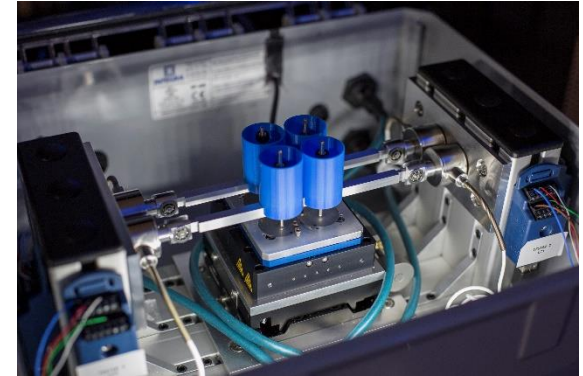
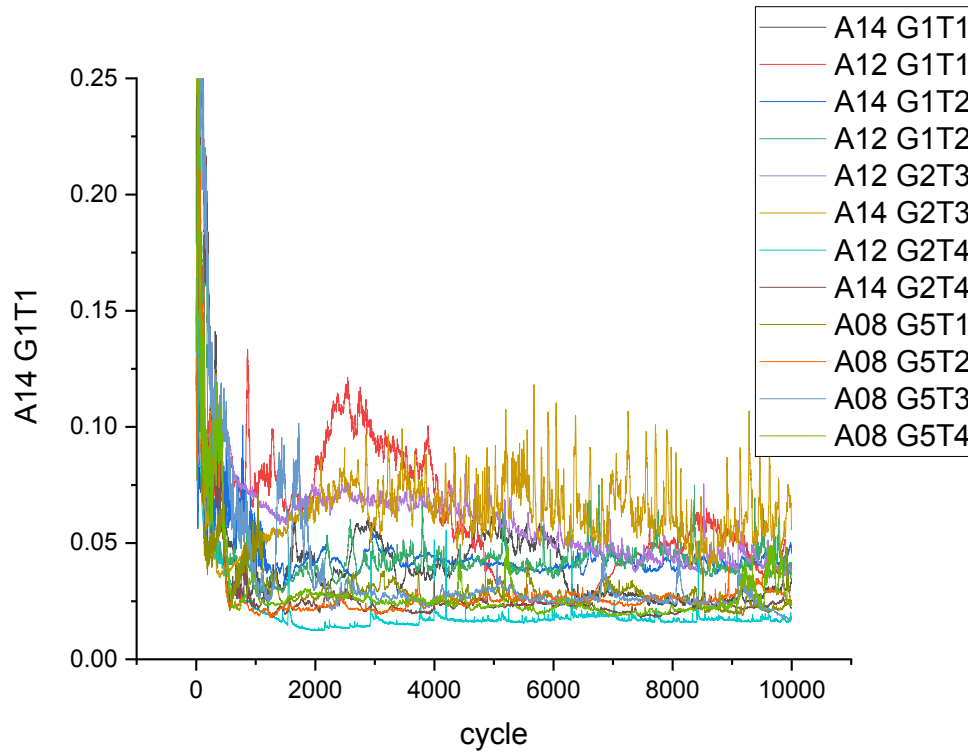
Variability in run-in (like MoS₂) can be a detriment to reliable operation

Brute Force – High Throughput Friction Testing



16X simultaneous testing capacity, multiple environments

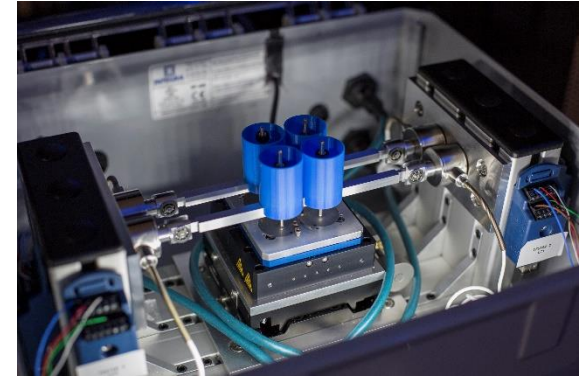
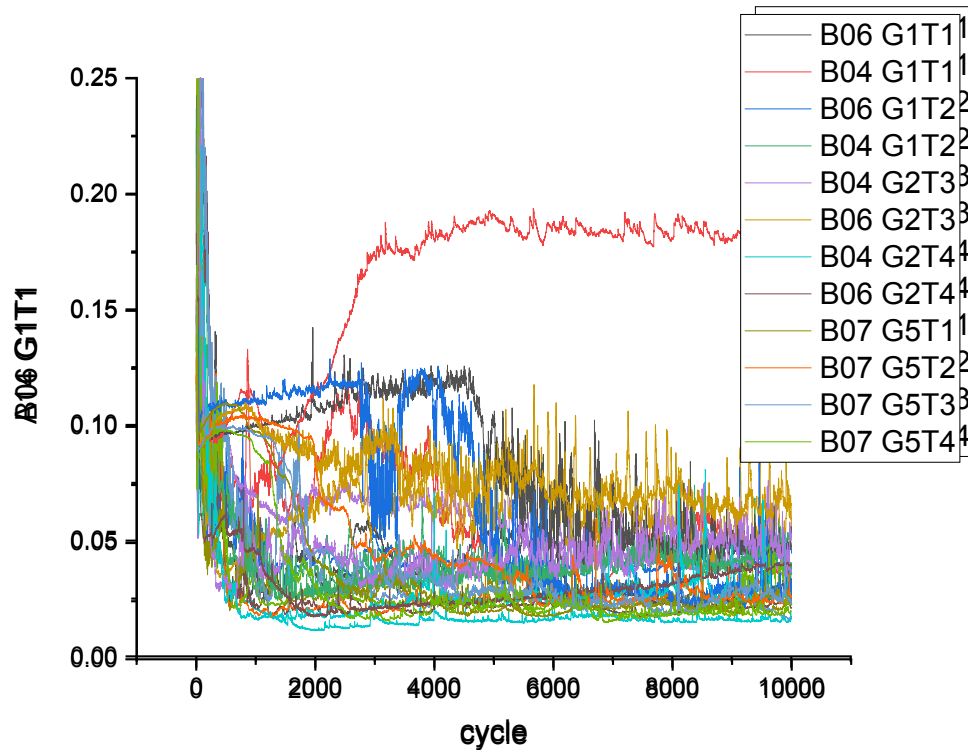
HTT Case Study - Composition



Test Specifics

- 5 DLC coatings; 4 samples each; 3-4 tests per sample @ 10,000 cycles
- 500 MPa (130 mN) max hertz contact pressure; 2mm stroke, 1 mm/s

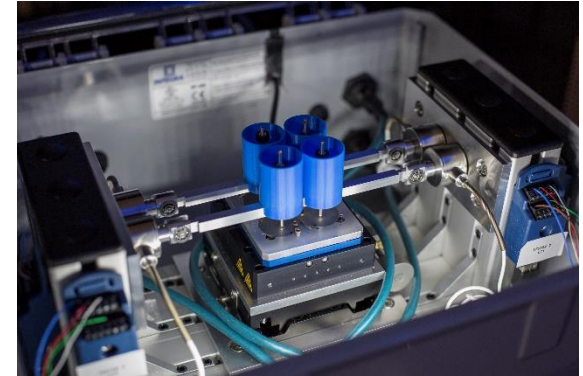
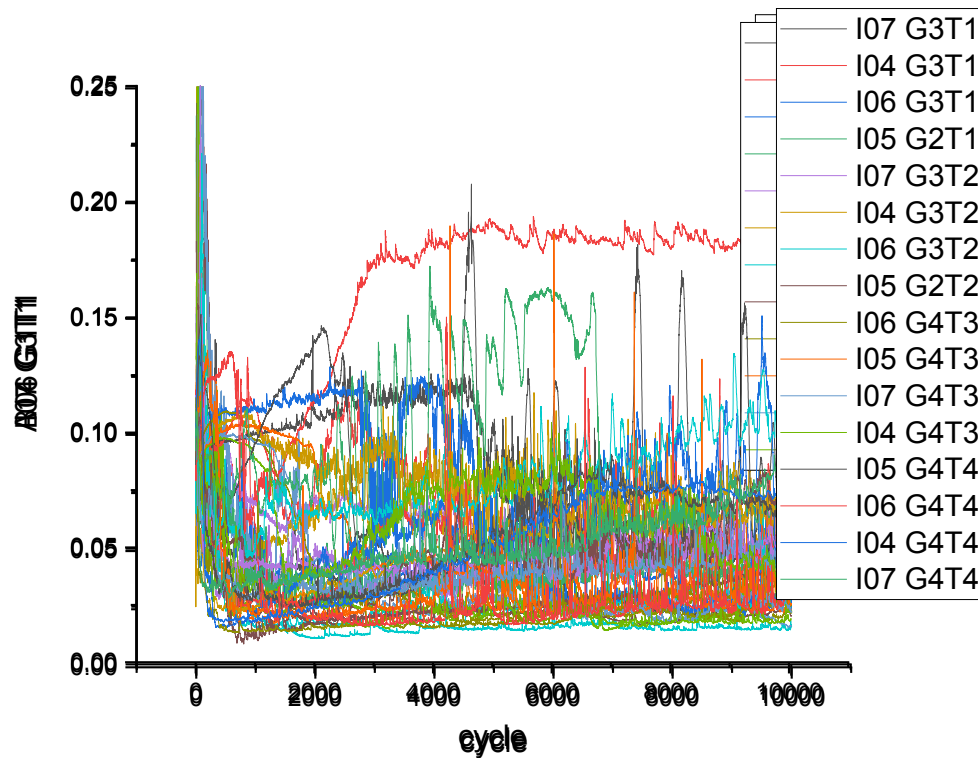
HTT Case Study - Composition



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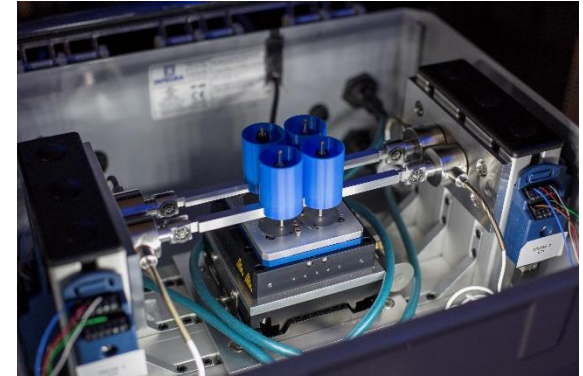
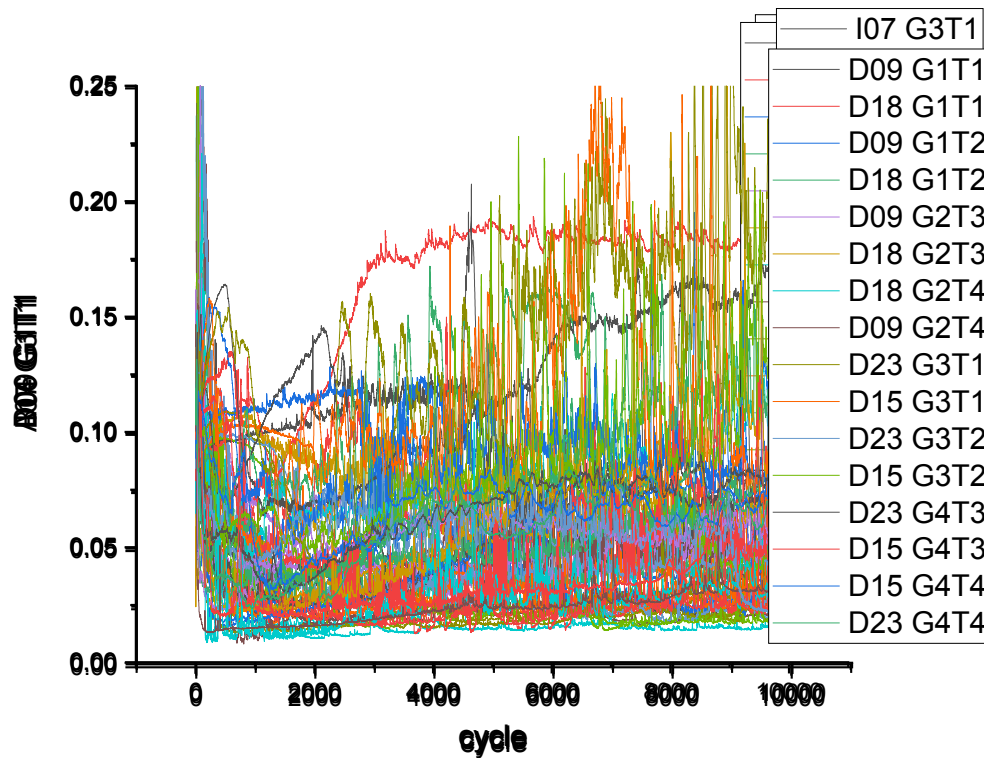
HTT Case Study - Composition



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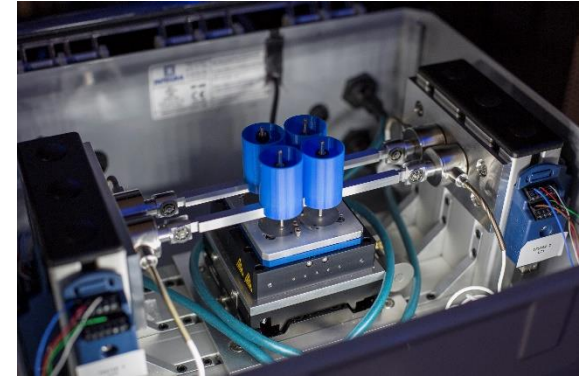
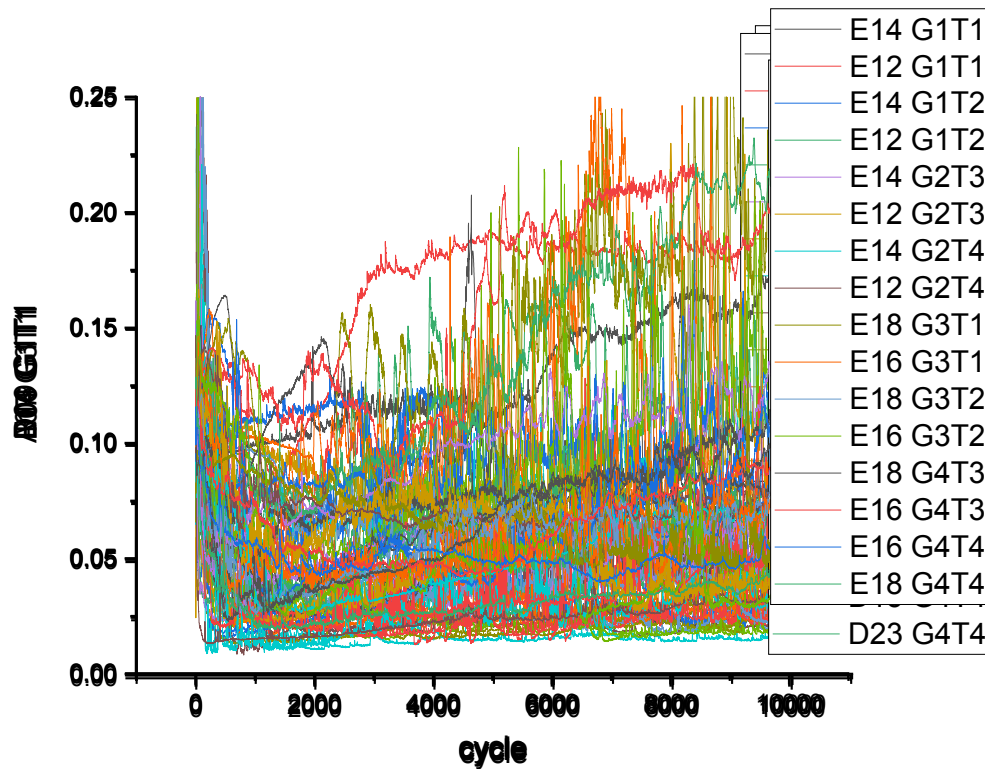
HTT Case Study - Composition



Test Specifics

- 5 DLC coatings; 4 samples each; 3-4 tests per sample @ 10,000 cycles
- 500 MPa (130 mN) max hertz contact pressure; 2mm stroke, 1 mm/s

HTT Case Study - Composition



Test Specifics

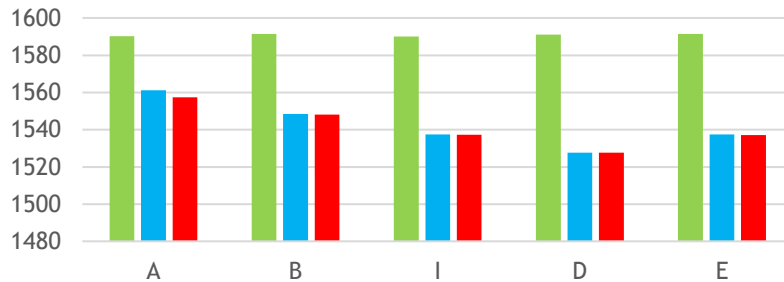
- 5 DLC coatings; 4 samples each; 3-4 tests per sample @ 10,000 cycles
- 500 MPa (130 mN) max hertz contact pressure; 1mm stroke, 1 mm/s

~75 experiments running 10K cycles each, taking under 1 week to finish

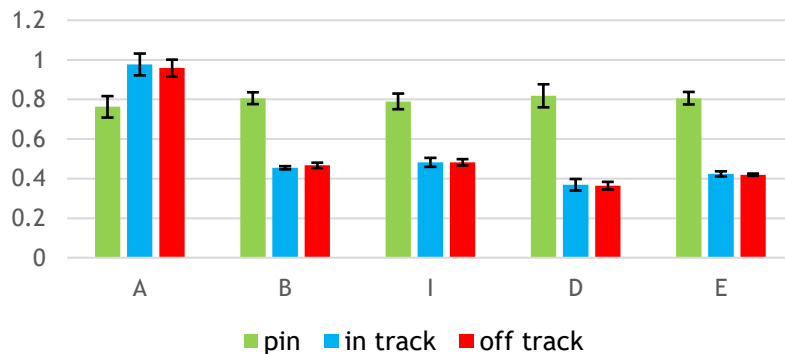
HTT Case Study - Results



Raman G Peak Position



Raman ID/IG Intensity Ratios

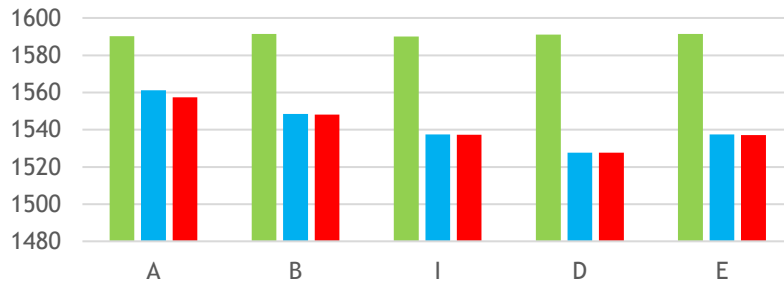


- In/out track Raman similar, exhibit differences unlike pin surface (unique from original film)

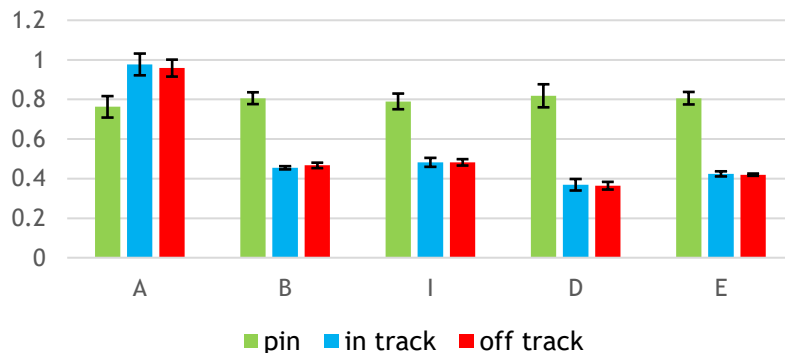
HTT Case Study - Results



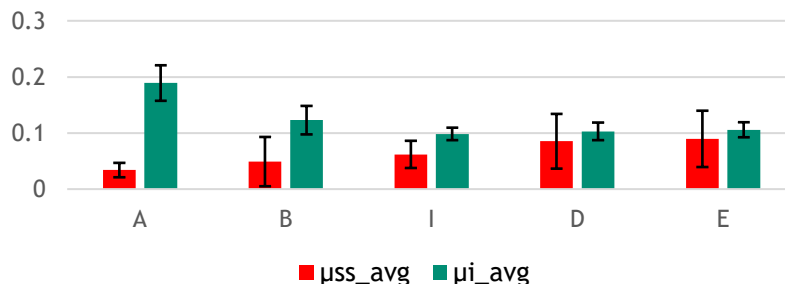
Raman G Peak Position



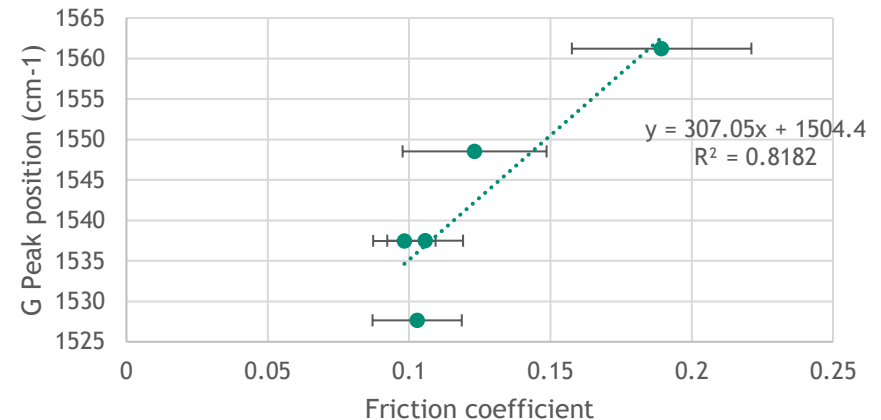
Raman ID/IG Intensity Ratios



Friction behavior



Initial Friction vs track G Peak Position

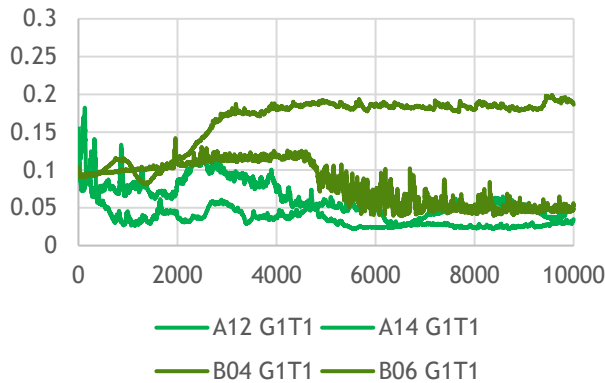


- In/out track Raman similar, exhibit differences unlike pin surface (unique from original film)
- Steady state and initial friction are inverse; correlate well with G peak pos & ID/IG of track

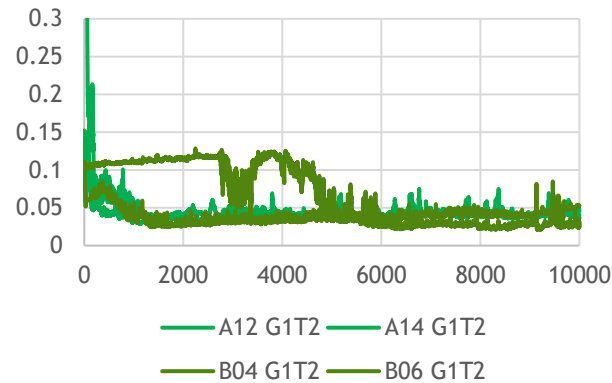
Effects of Purge Time on Variability



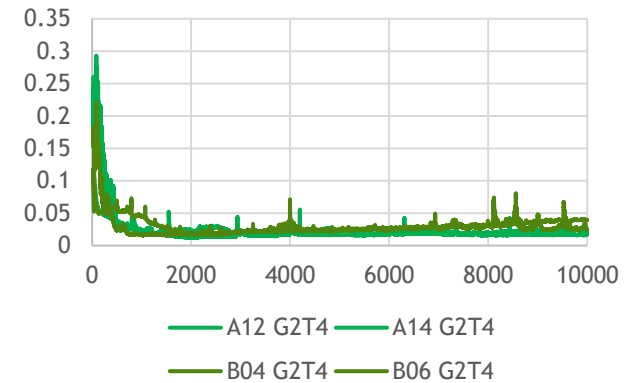
Group 1, Test 1, Module 1 - Purge Time 1hr



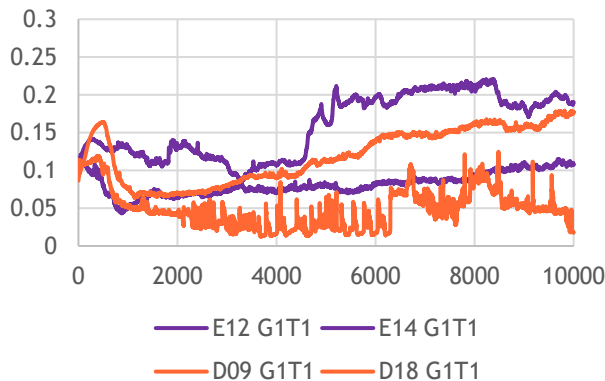
Group 1, Test 2, Module 1 - Purge Time 8hr



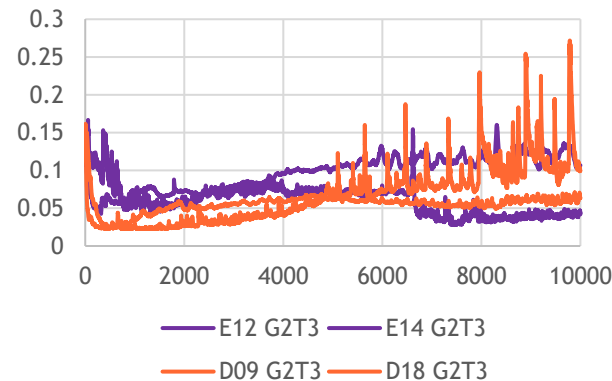
Group 2, Test 4, Module 2 - Purge Time 96hr



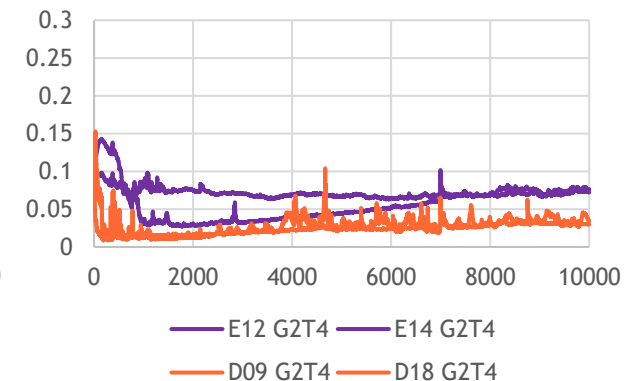
Group 1, Test 1, Module 2 - Purge Time 1hr



Group 2, Test 3, Module 1 - Purge Time 8hr



Group 2, Test 4, Module 1 - Purge Time 96hr

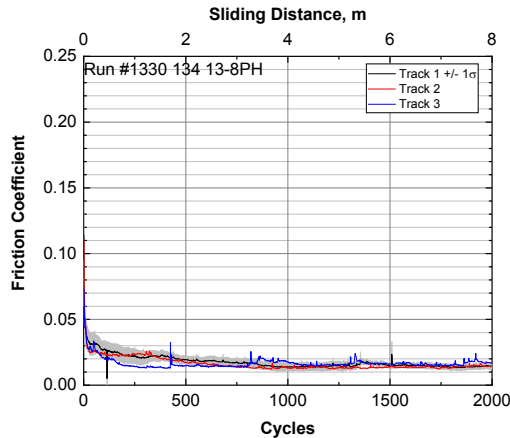


* All labels are friction coefficient vs cycle
** 1 hour purges target <20 ppm O₂/H₂O

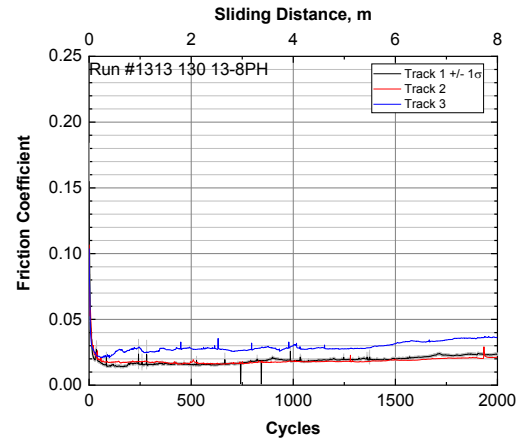
Role of Surface Termination



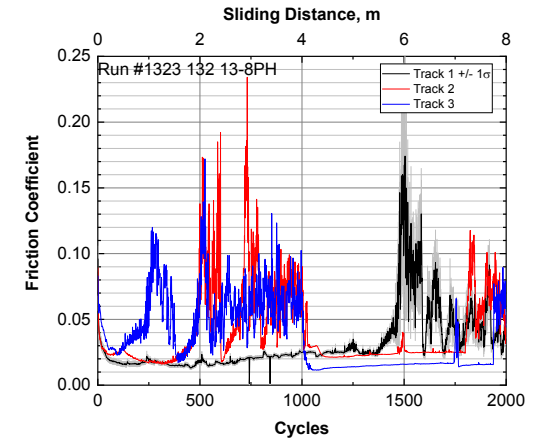
Baseline
(Run 5)



Ramp Down C6H12, Ramp Up Ar
(Run 2)



Ramp RF Power Down
(Run 4)

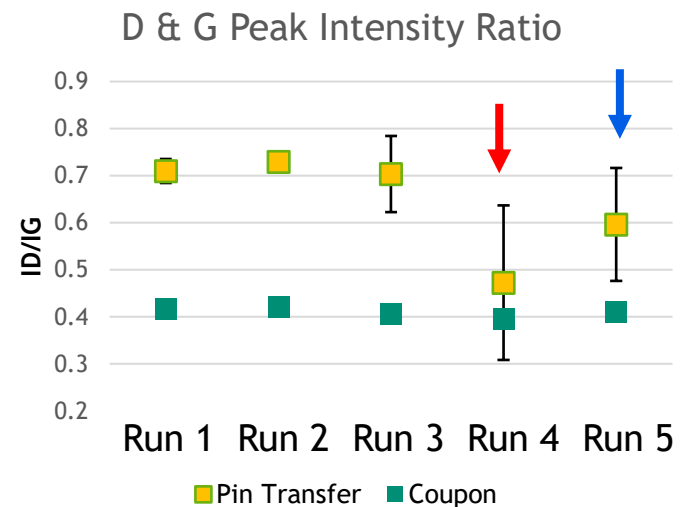
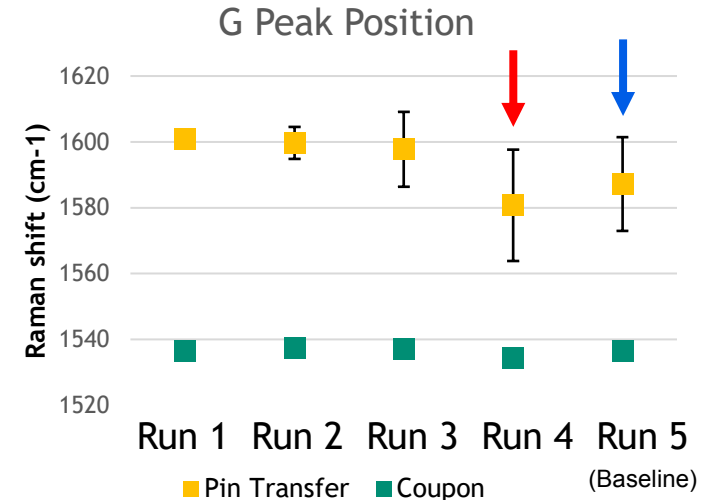


- All deposition runs same as baseline process, except for how the process ended
 - baseline involves shutting down gas precursors and RF power simultaneously
- The friction behavior was clearly modified by changing the shut-down process
 - there may be a decrease in run-in for some processes compared to baseline – more data (high throughput testing) will be used to evaluate significance

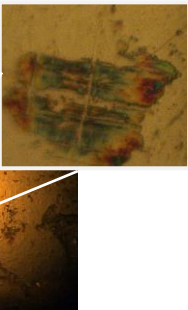
Surface Termination - Raman



- As expected, spectra from coupons remained similar
- Spectra taken on pin transfer exhibited differences from baseline (Run 5) in all cases:
 - RF power ramp down (Run 4) – lower ID/IG and G Peak position; also exhibited highly erratic friction behavior
 - Runs 1,2,3 – higher ID/IG and G Peak position; exhibited similar friction behavior to baseline
- Results suggest that changes in friction behavior are linked to process changes that alter interfacial chemistry



Example
Pin transfer



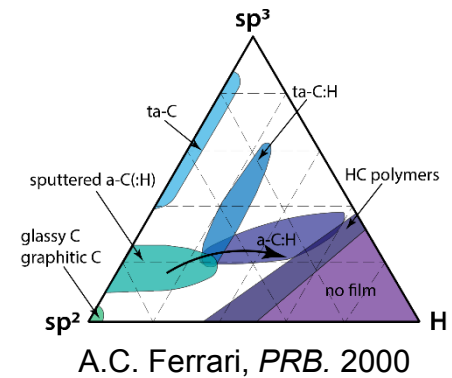
Run ID	Process
1	Ar ion etch
2	Ramp Down C ₆ H ₁₂ , Ramp Up Ar
3	Ramp Down C ₆ H ₁₂
4	Ramp RF Power Down
5	Baseline Process

Key Take-Aways

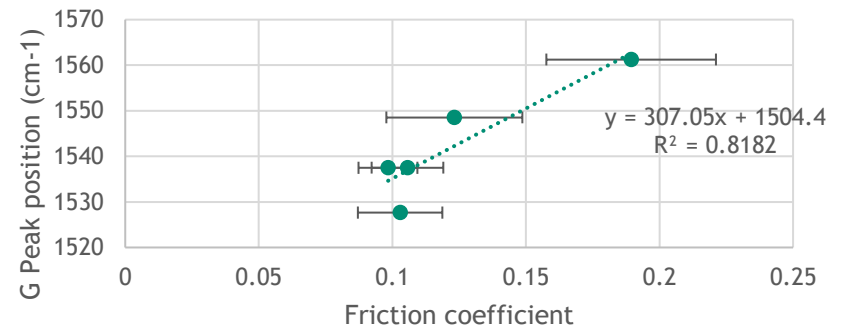
- Confirmed link between composition (via Raman/HFTF) and friction behavior for different vendors
- Surface termination simple route to change friction behavior
- Can utilize these relationships to design better coatings gear to run-in

Ongoing / Future Work

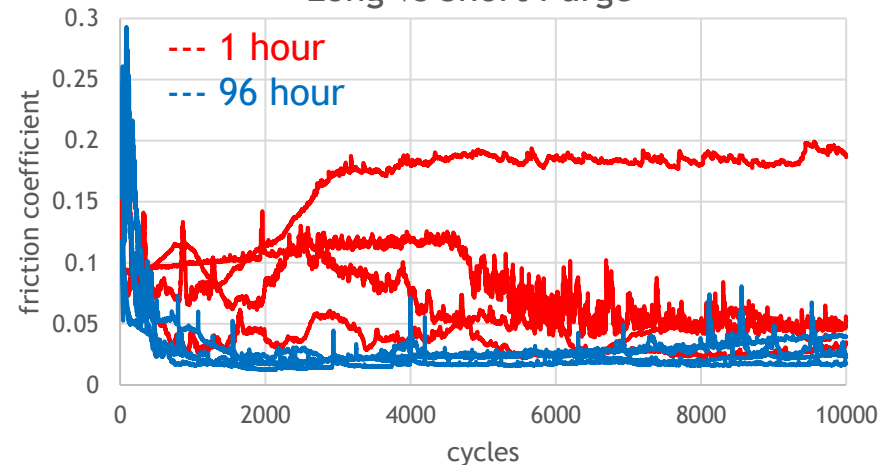
- Additional In situ testing (Raman/NEXAFS) for cycle resolved changes at surface
- What factors during purge change (ambient RGA)



Initial Friction vs track G Peak Position



Long vs Short Purge

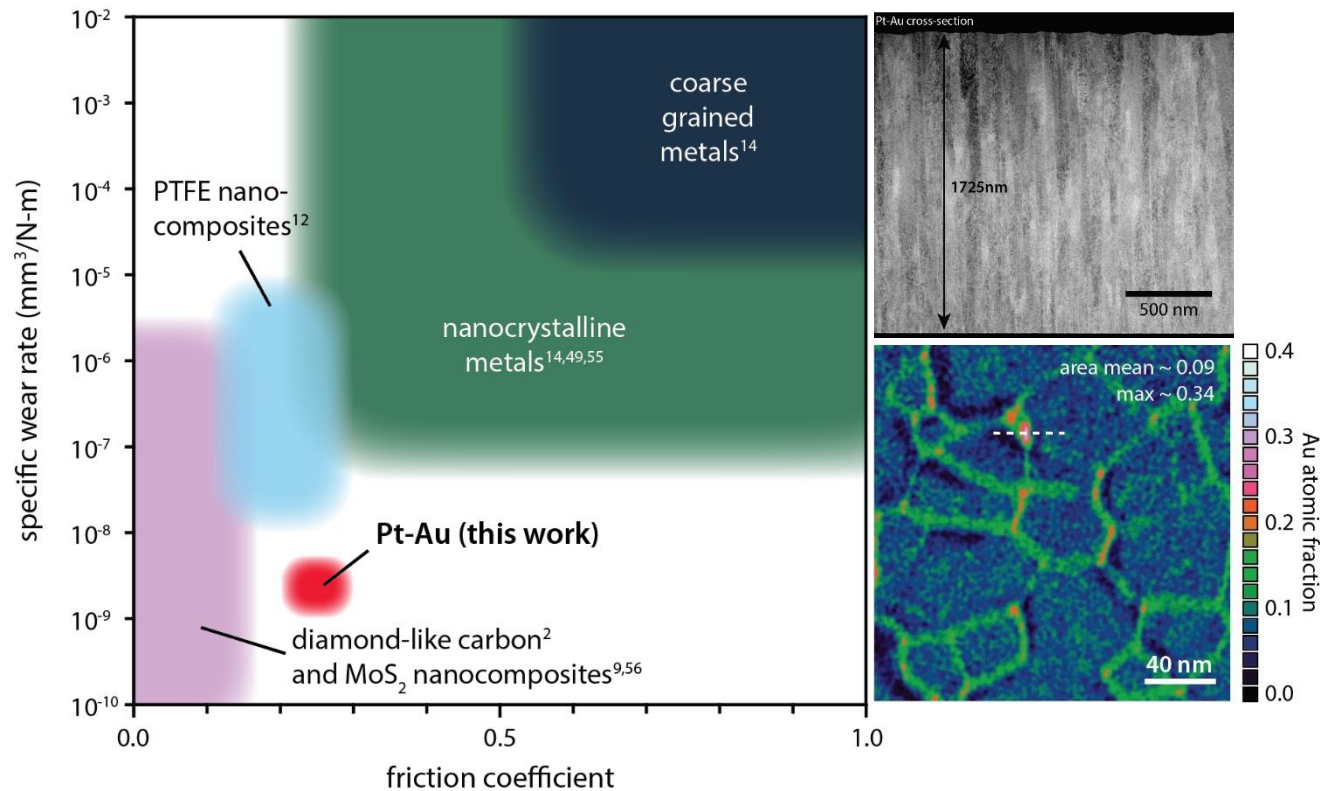




Self-lubricating in situ carbon films (SLIC)

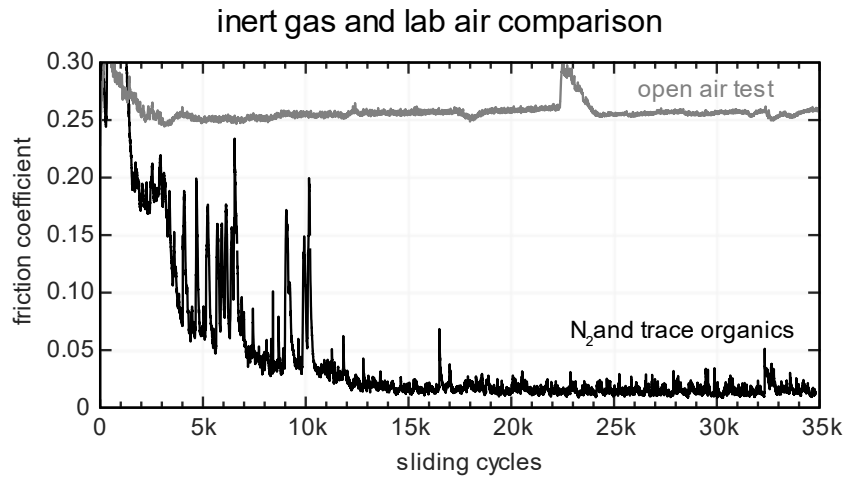
A decorative horizontal bar with a series of colored segments (blue, yellow, green, pink, purple, orange, etc.) is positioned below the title.

Discovery: Ultra-Low Wear Pt-Au



- Byproduct of LDRD on development of stable, ultra-nanocrystalline alloys
- Most tests run in air – needed to check performance in inert environments...

Unexpected Tribocatalysis

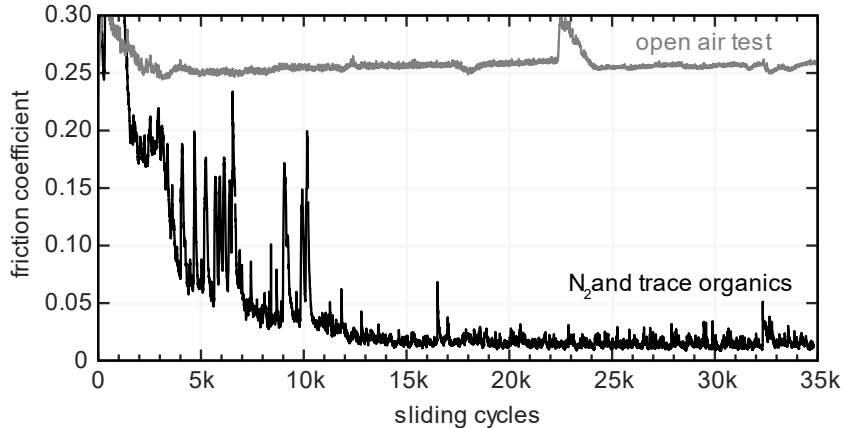


- Testing in inert environments lowers friction?

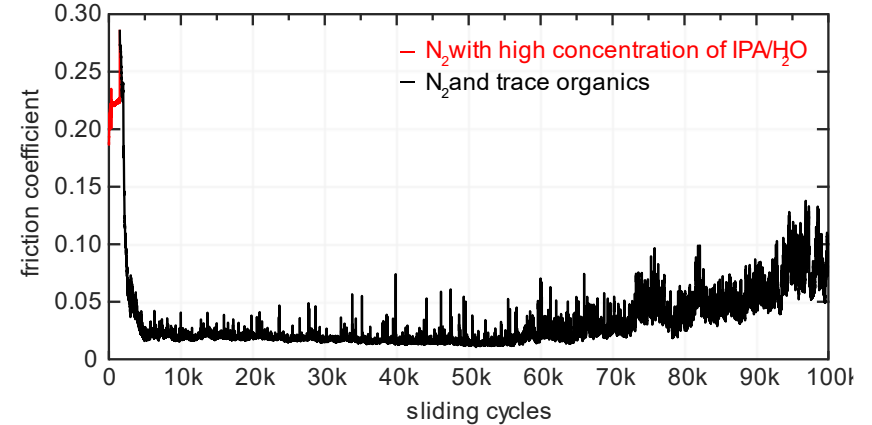
Unexpected Tribocatalysis



inert gas and lab air comparison



inert gas with water and alcohol vapor

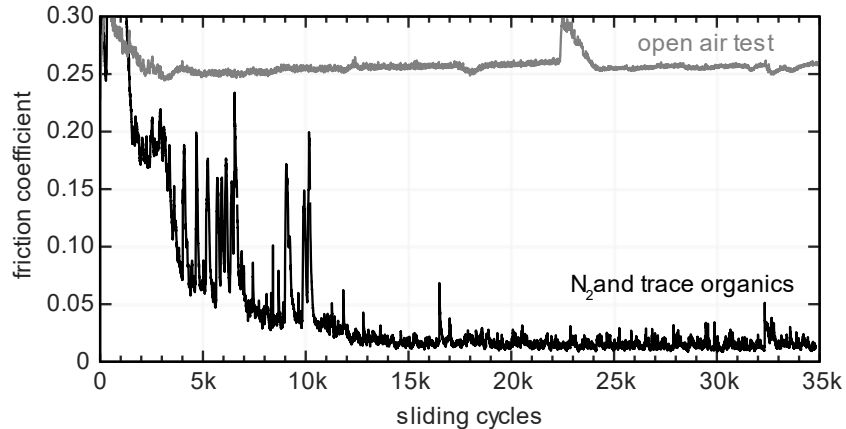


- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop...

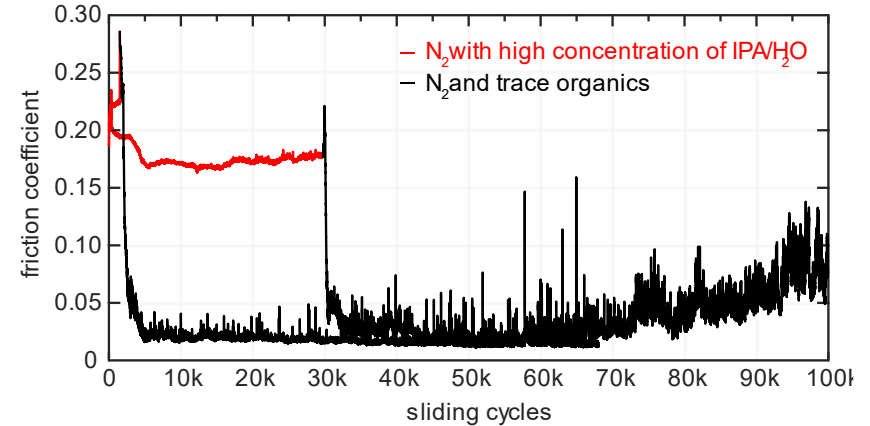
Unexpected Tribocatalysis



inert gas and lab air comparison



inert gas with water and alcohol vapor

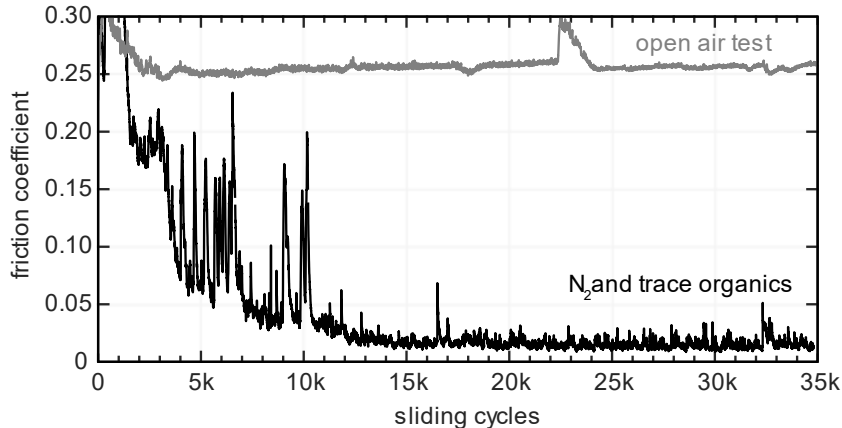


- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop... and prolongs it

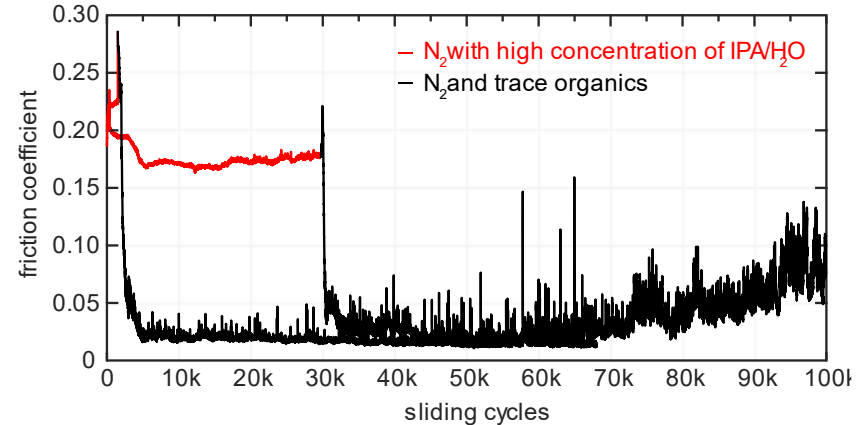
Unexpected Tribocatalysis



inert gas and lab air comparison

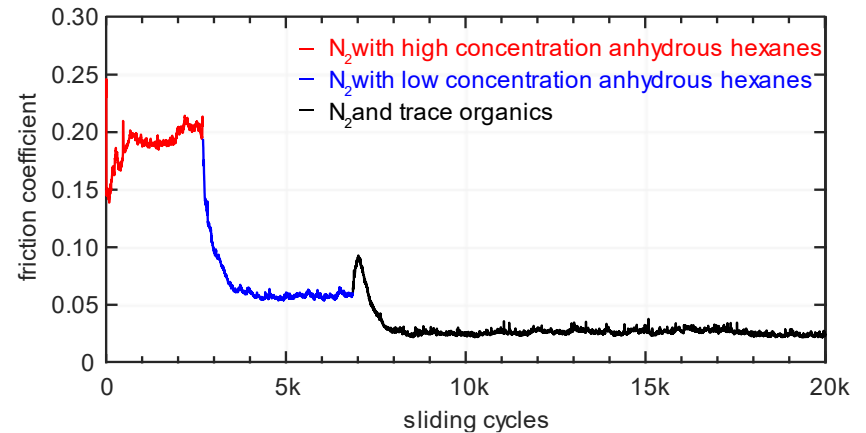


inert gas with water and alcohol vapor



- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop... and prolongs it
- Any amount of anhydrous hexanes increased friction, with higher/lower friction at higher/lower concentrations
- Unclear what role water/oxygen play

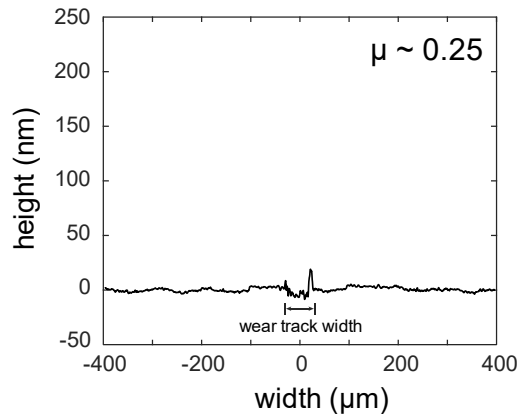
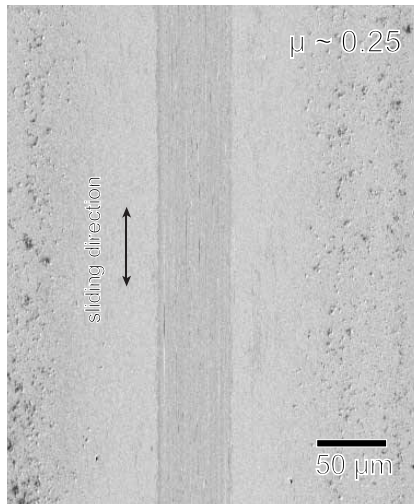
inert gas with anhydrous hexanes



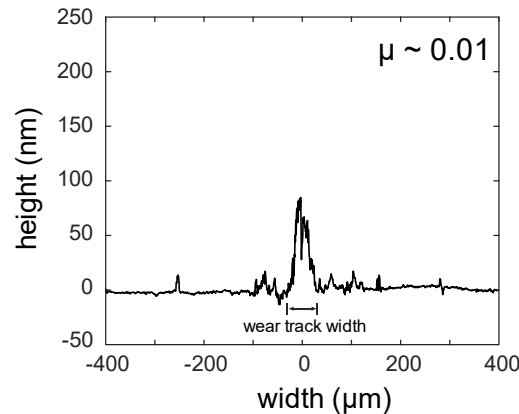
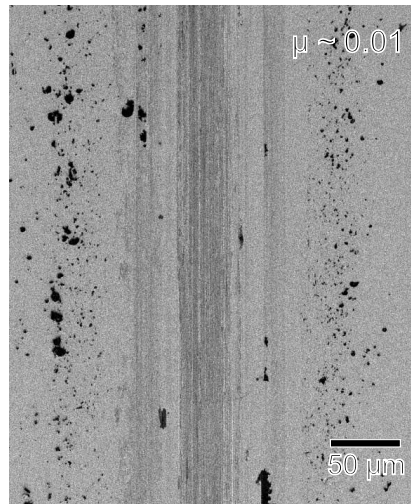
Accumulation is Key



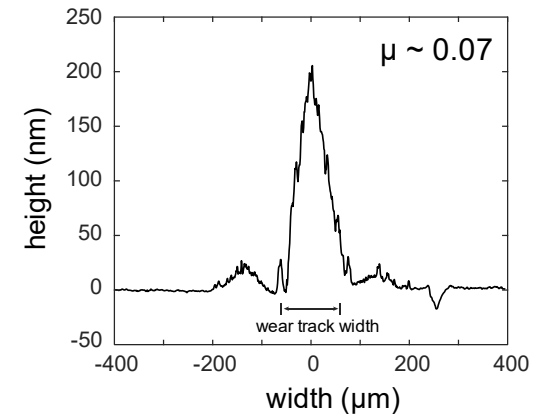
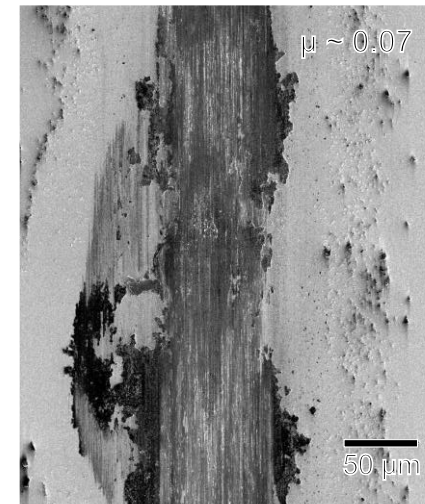
A. lab air



B. N₂ and trace organics



C. N₂ and high concentration IPA/H₂O

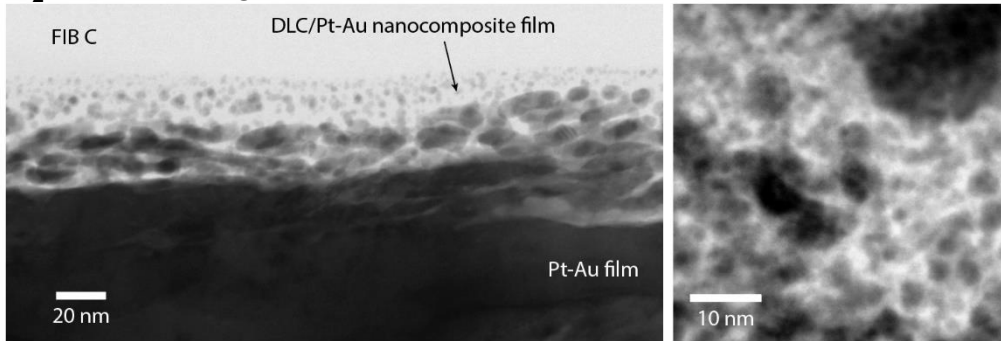


- Concentration also affects film growth in wear scar
- Highest concentrations produces thick films unable to reach low friction state

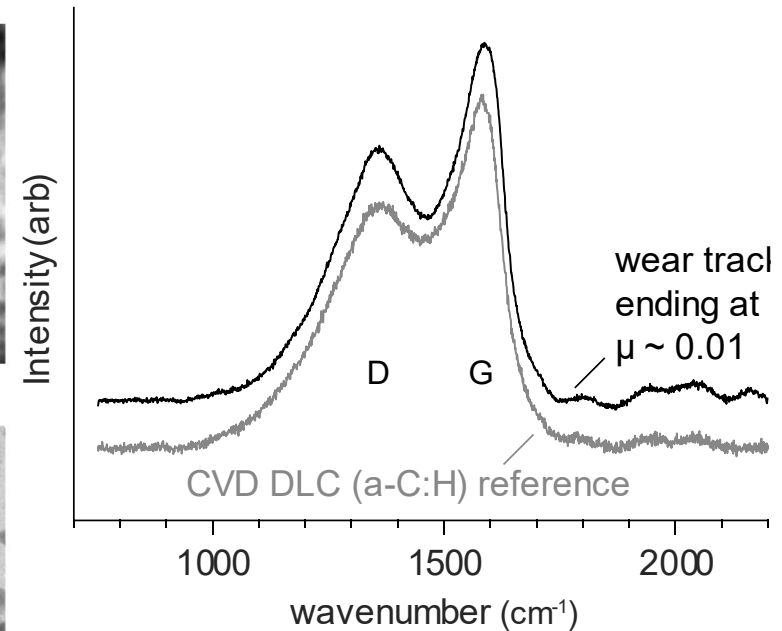
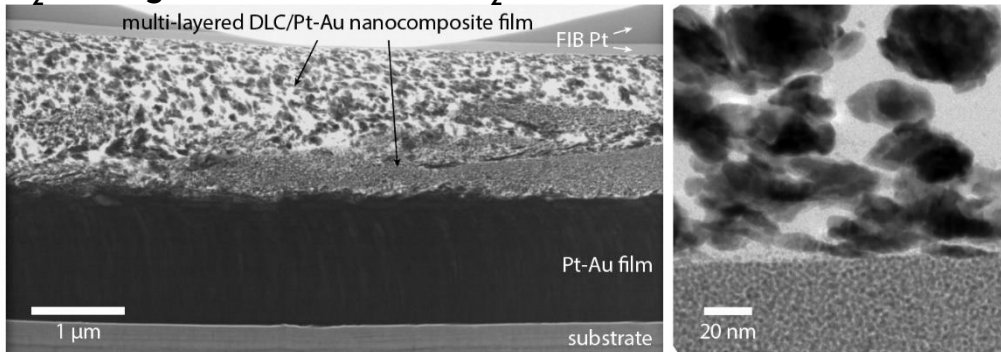
A Tribo-polymeric Nanocomposite



N₂ and trace organics



N₂ and high concentration IPA/H₂O

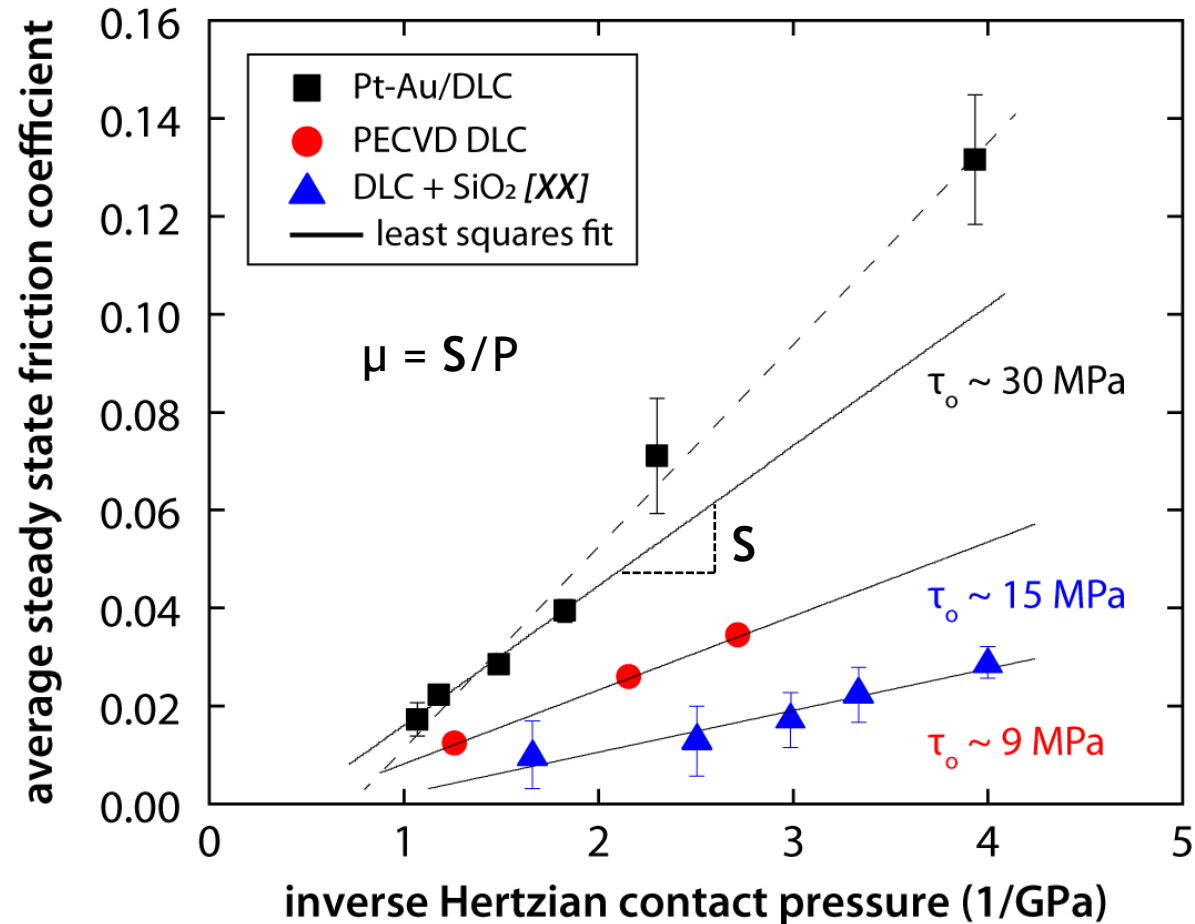


- Films are actually composite of Pt-Au nanoparticles and DLC-like carbon, confirmed by TEM & Raman
- High concentrations exhibit phases of larger, less mixed/layered particles, possibly limiting mixing & Pt interaction at surface

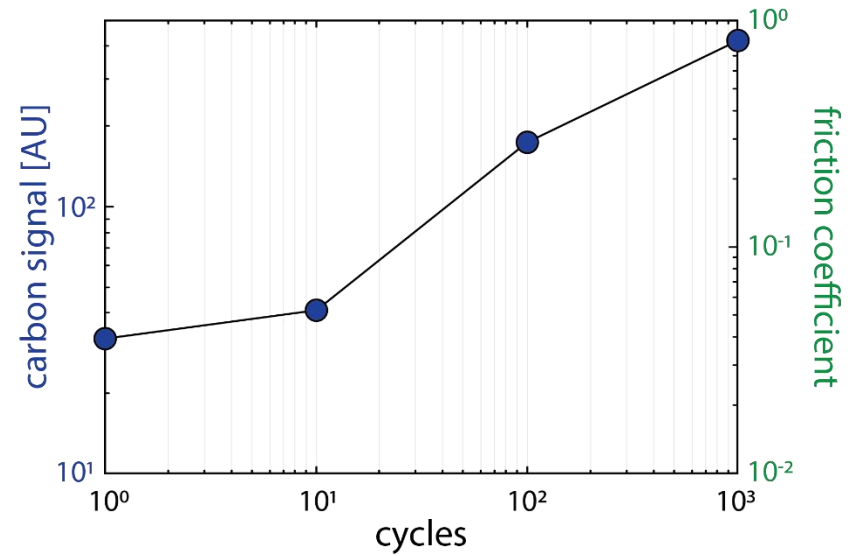
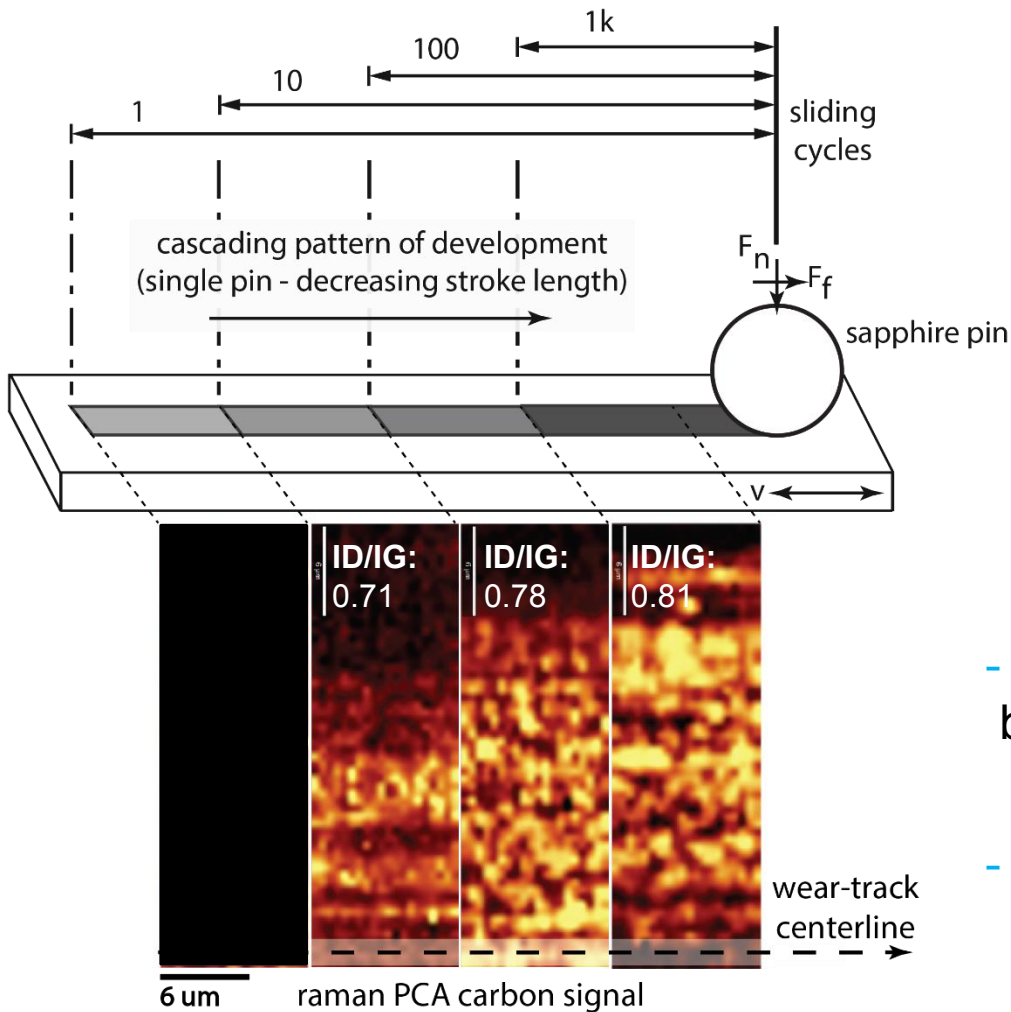
Qualifying Shear Strength



- Can derive shear strengths from Hertzian contact model
- Shear strength comparable to commercially available
- Discrepancies may be due to composite nature of film and lower hydrogenation (20 vs 40%)

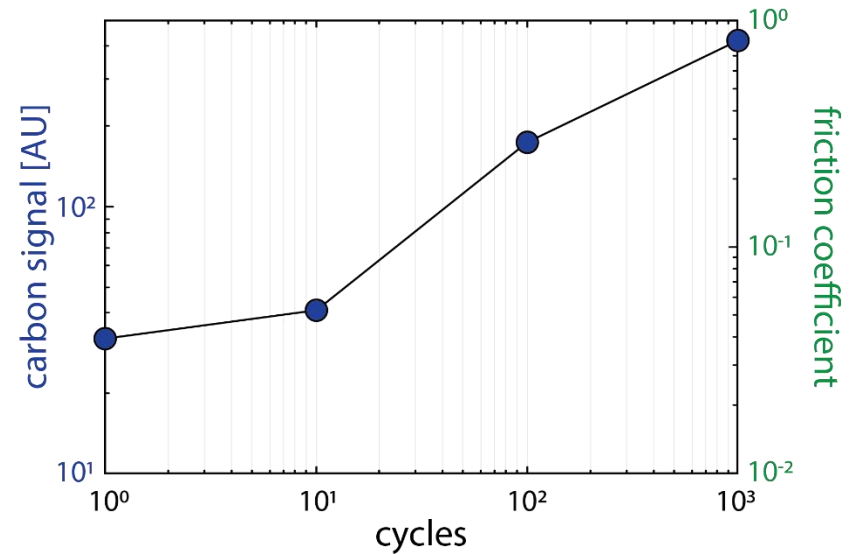
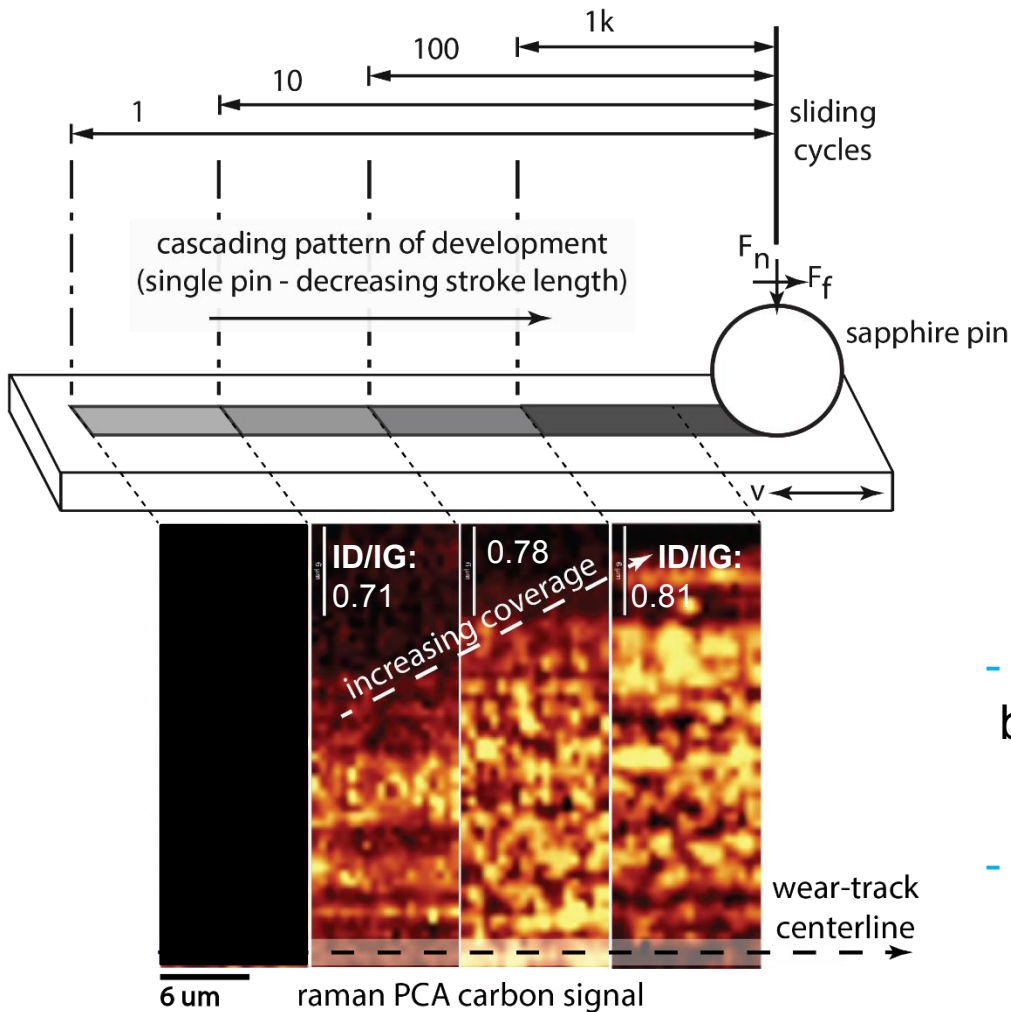


Time Dependent Formation



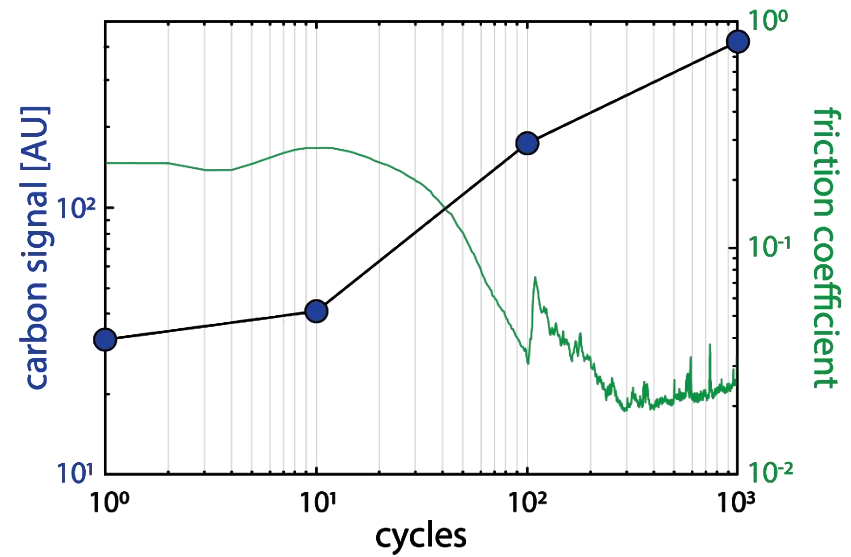
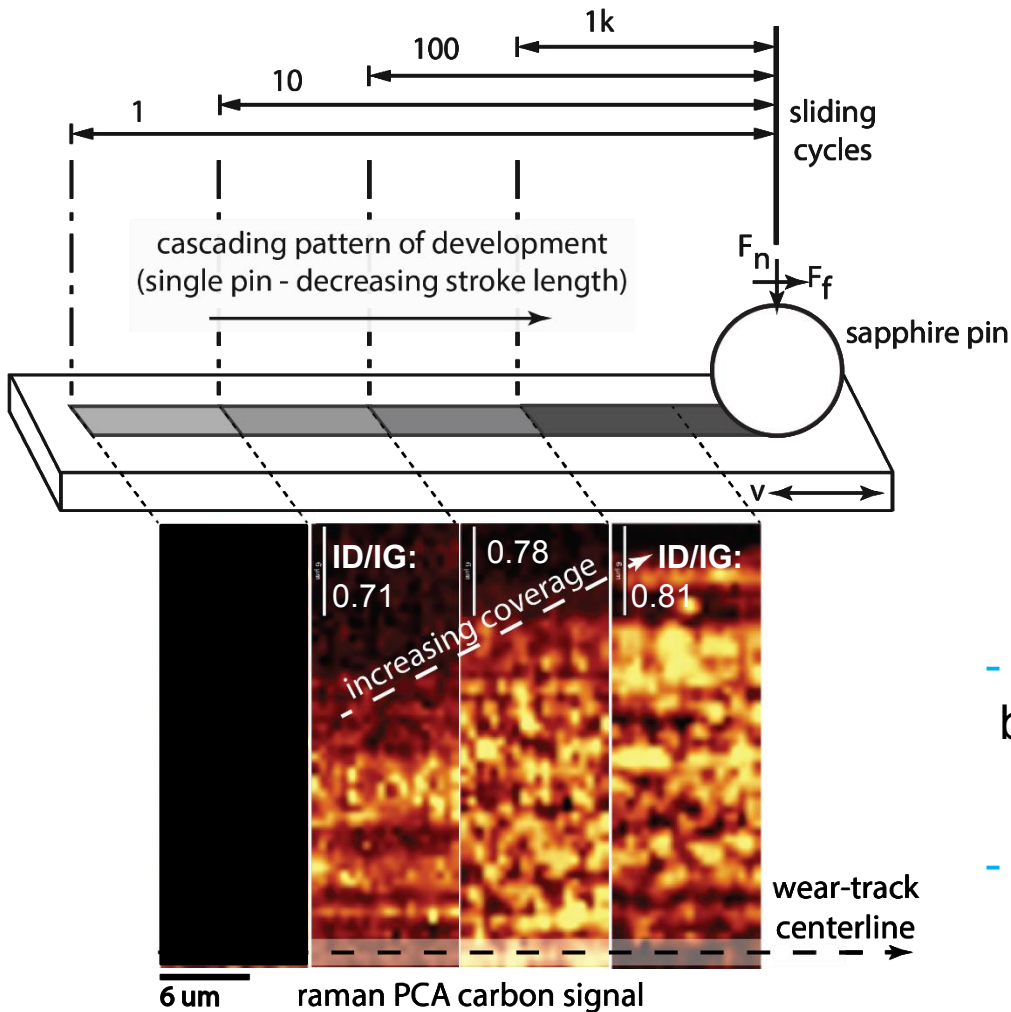
- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
 - Stronger carbon signals in wear track

Time Dependent Formation

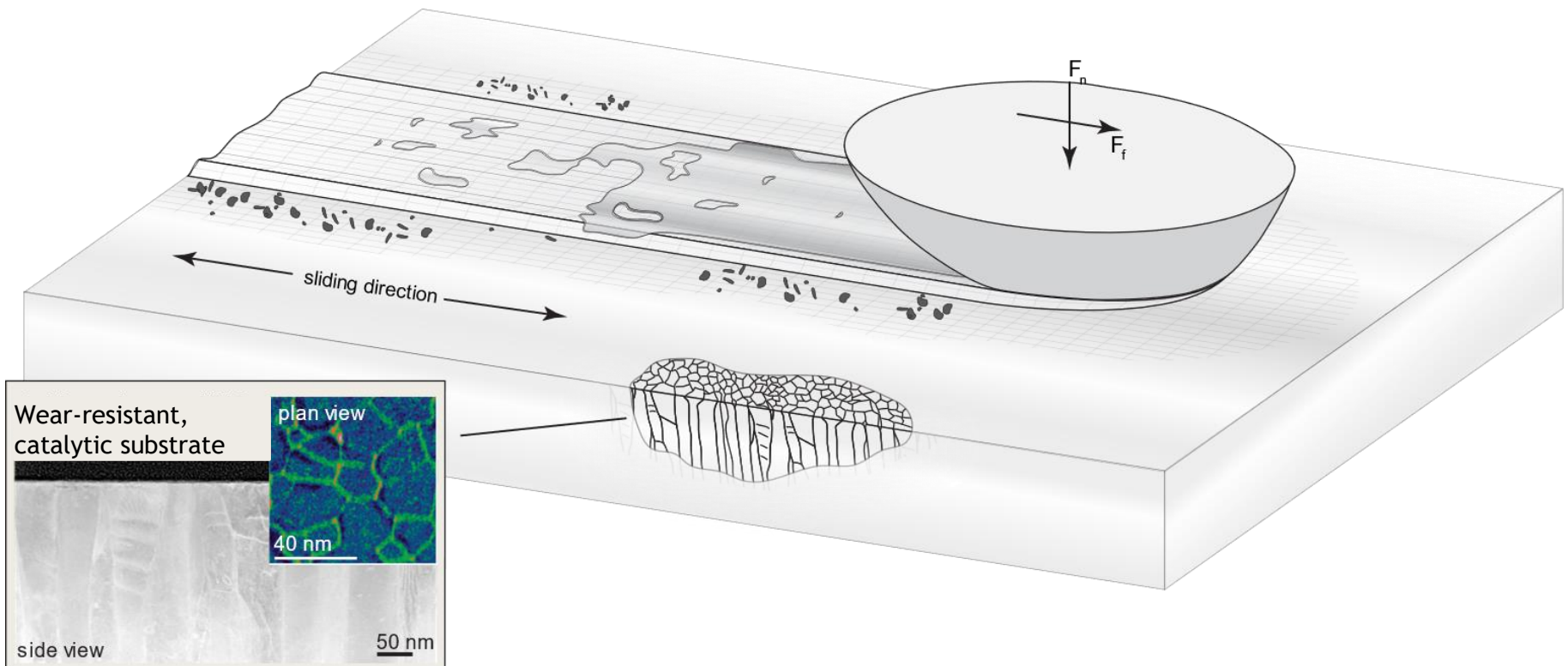


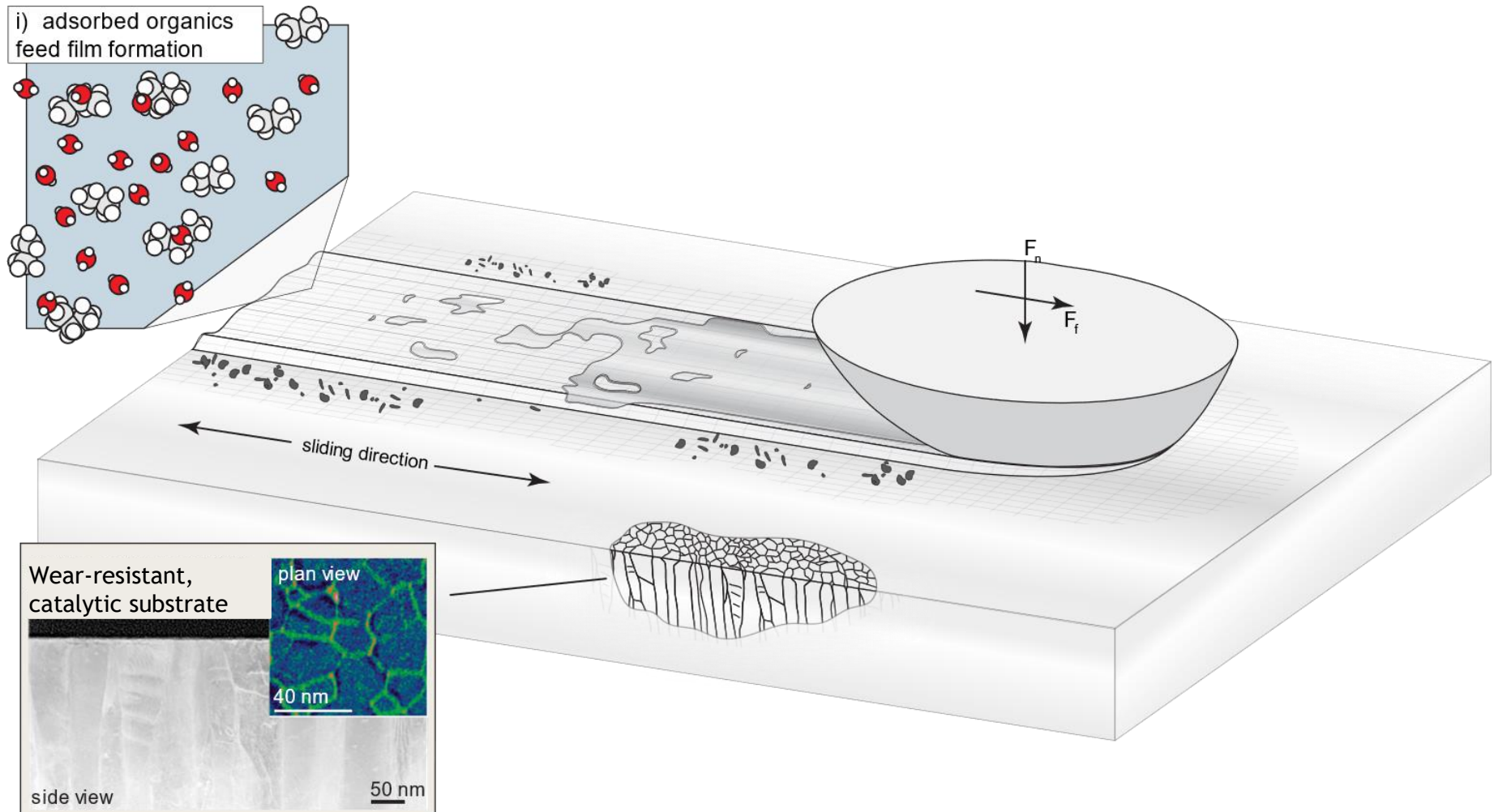
- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
 - Stronger carbon signals in wear track
 - Higher coverage in wear scar

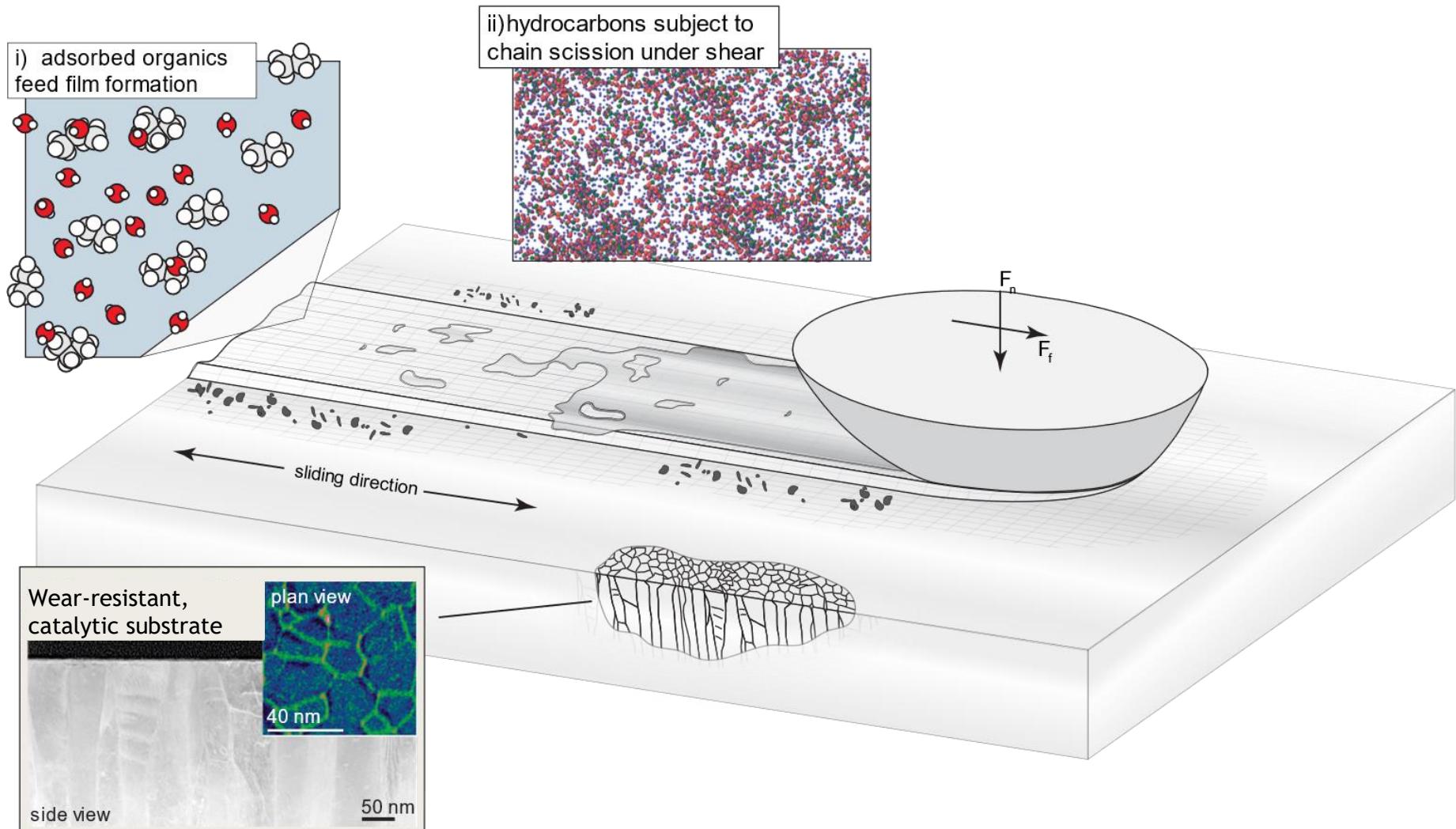
Time Dependent Formation

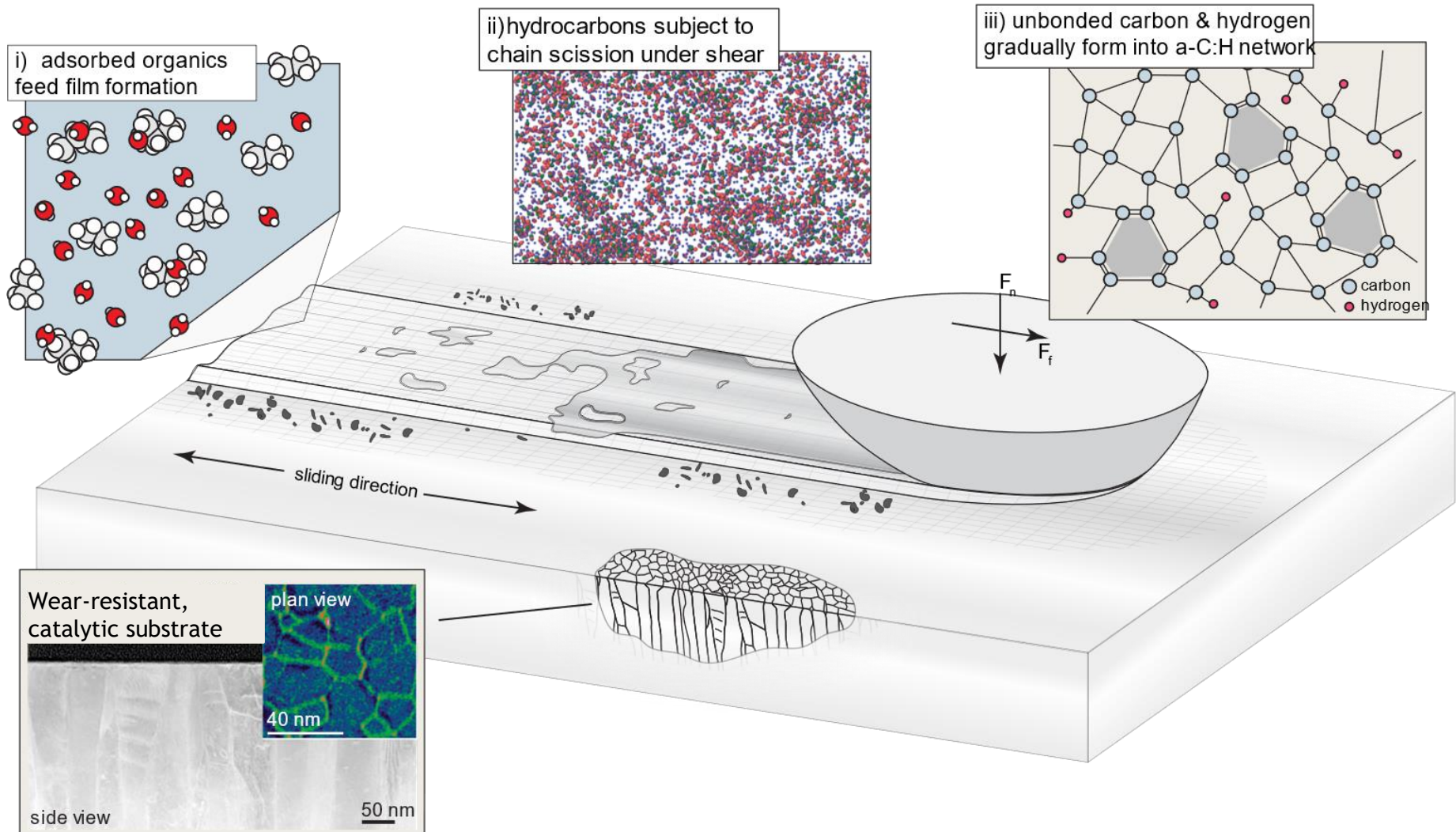


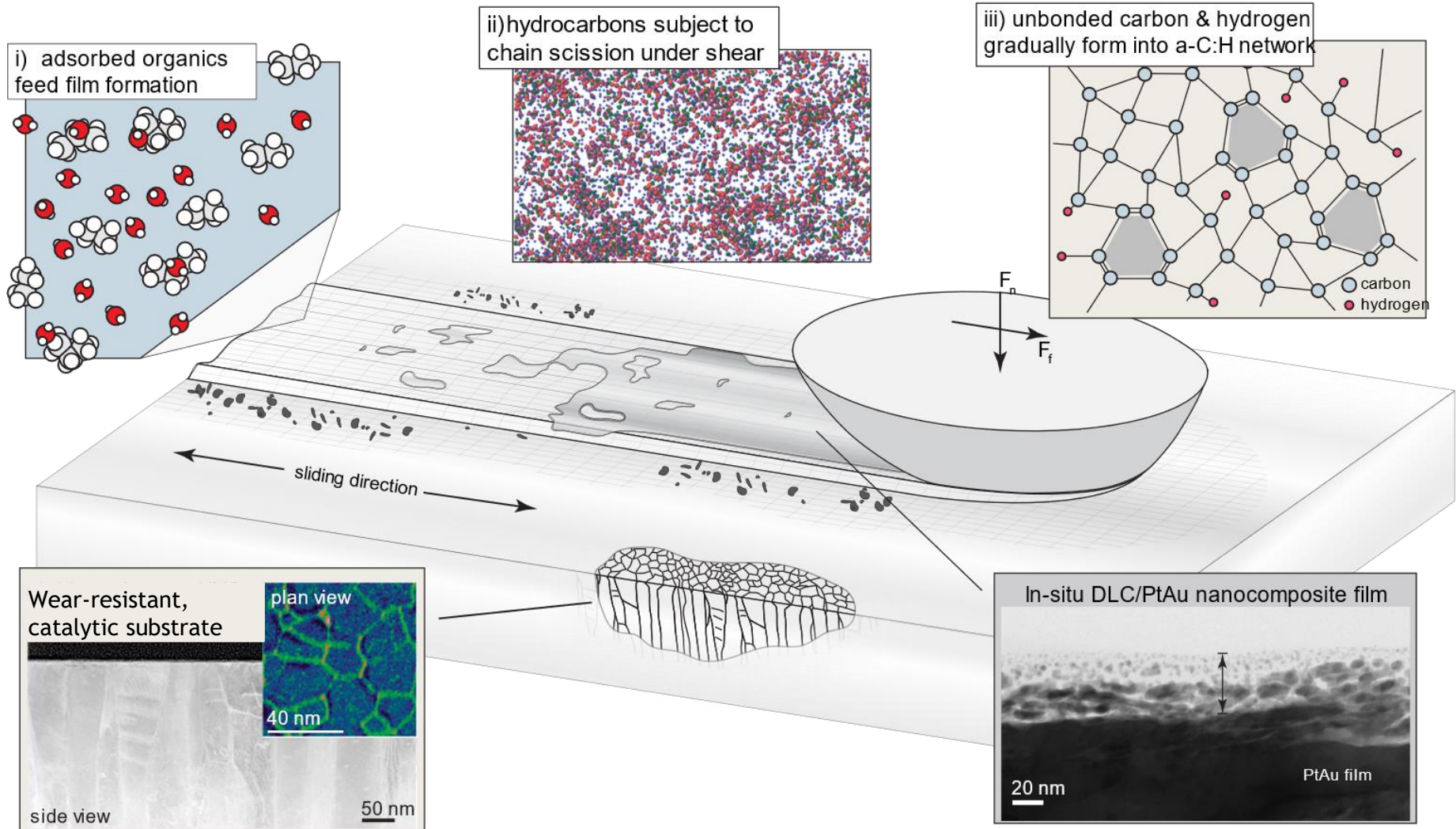
- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
 - Stronger carbon signals in wear track
 - Higher coverage in wear scar
 - Decreasing friction coefficient











Key Take-Aways

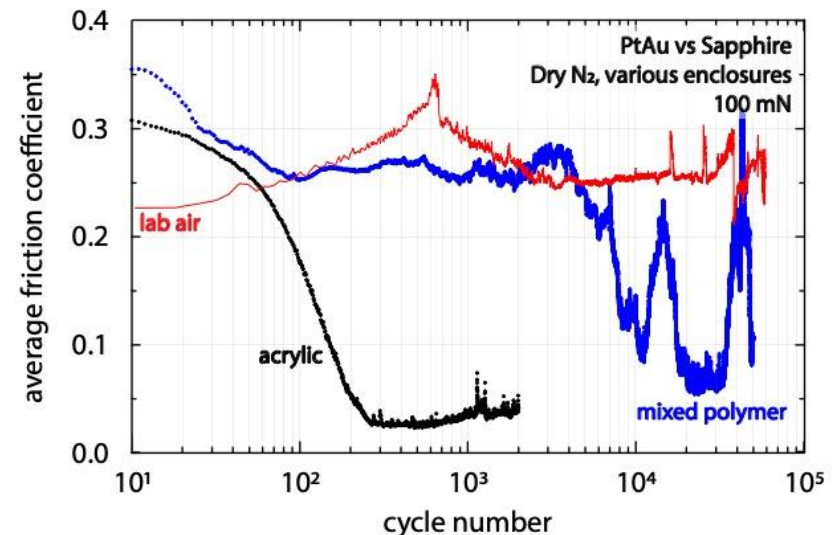
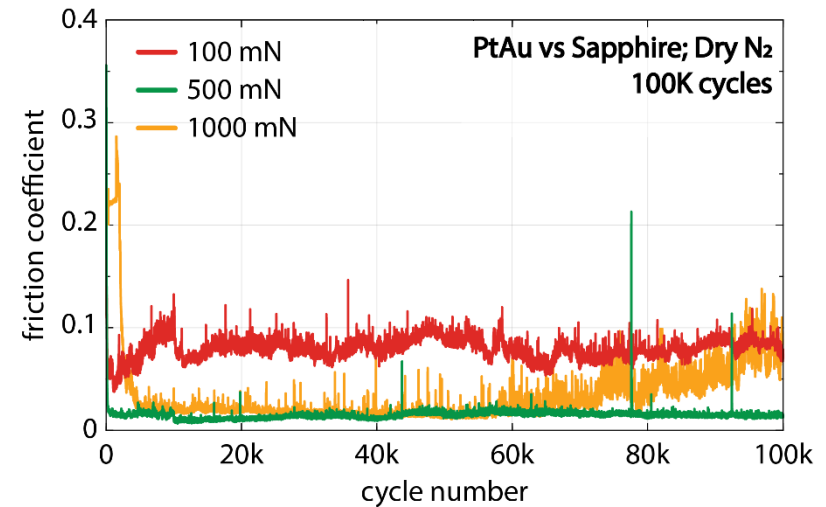


Outcomes

- Trace concentrations of hydrocarbons enable formation of SLIC, a DLC-like tribofilm
- SLIC films exhibit stress/cycle dependent formation (& temperature in ethanol!)

Ongoing / Future Work

- Understanding of underlying mechanisms behind competing stress, time & wear
- What do the resulting microstructures give us in terms of friction, wear or conductivity?
- What is the ideal composition or concentration of species in the environment for formation?
- Can we utilize as deposition method?



Acknowledgements



Sandia

- **Michael Dugger** MoS₂ Aging studies
- **Michael Chandross, Scott Bobbitt** MoS₂ MD/DFT
- **Frank DeIRio** AFM/KPFM on MoS₂
- **Brendan Nation, JW Johnson, Morgan Jones, Brian Wisler** Experimental Support
- **Taisuke Ohta** XPEEM defect characterization on MoS₂
- **Prof. Thomas Beechem** DLC characterization, in-situ Raman capability development



- **Florida State University Prof Brandon Krick, Tomas Babuska** in vacuo oxidation studies of MoS₂

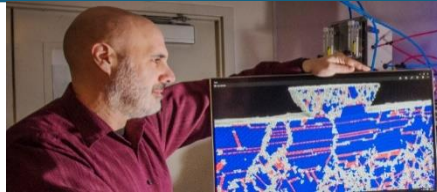
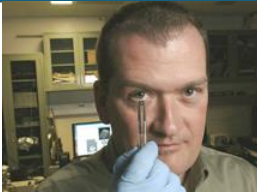
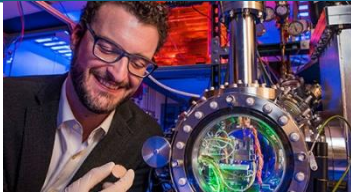
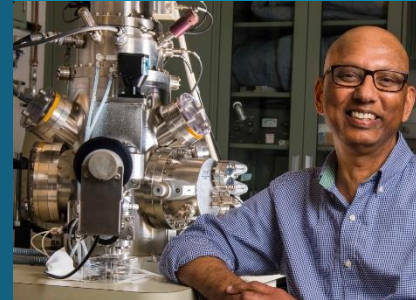
- **UT Austin Prof Filippo Mangolini, Robert Chrostowski** NEXAFS/APPEX experiments on MoS₂/DLC



TEXAS
The University of Texas at Austin



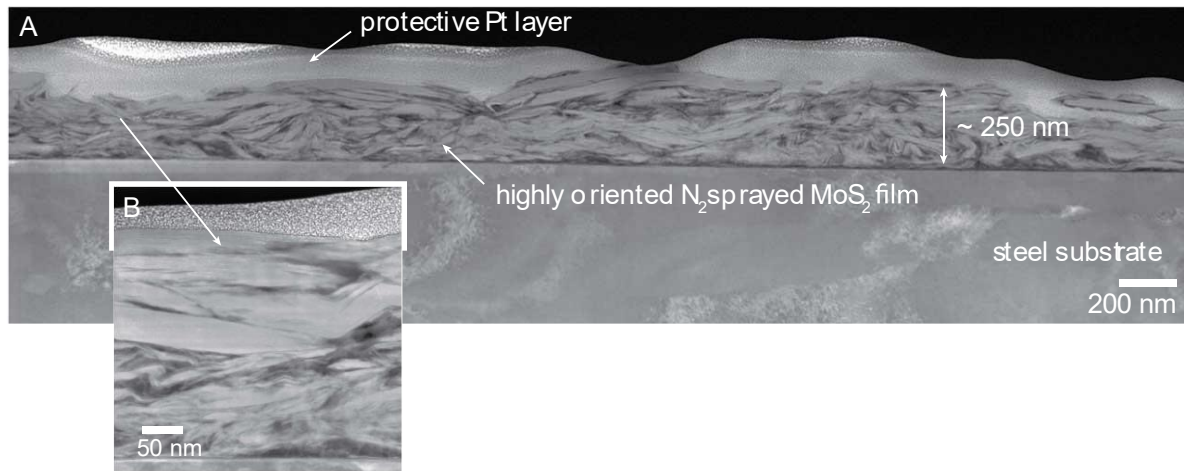
Thank You



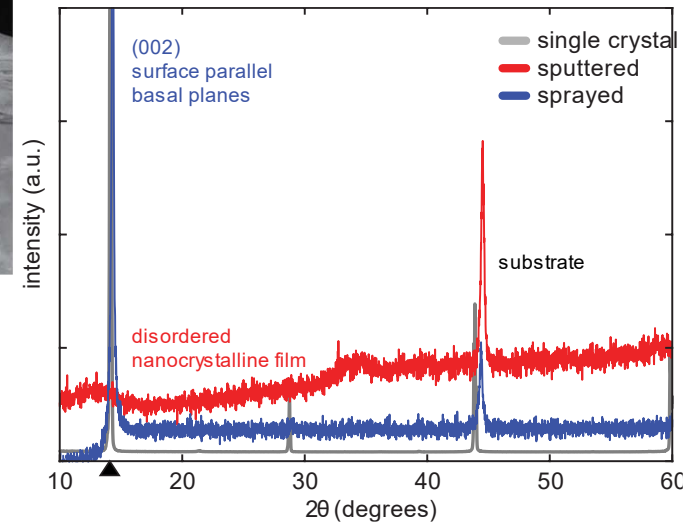
Questions?

APPENDIX

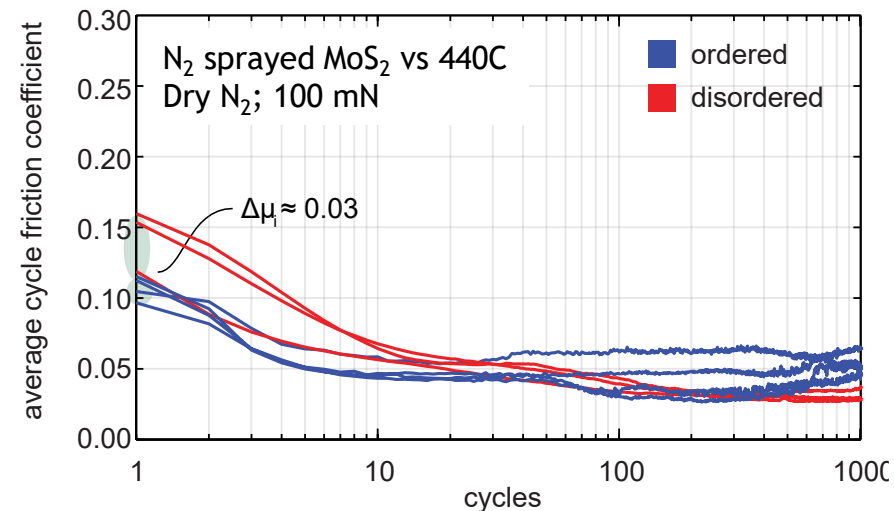




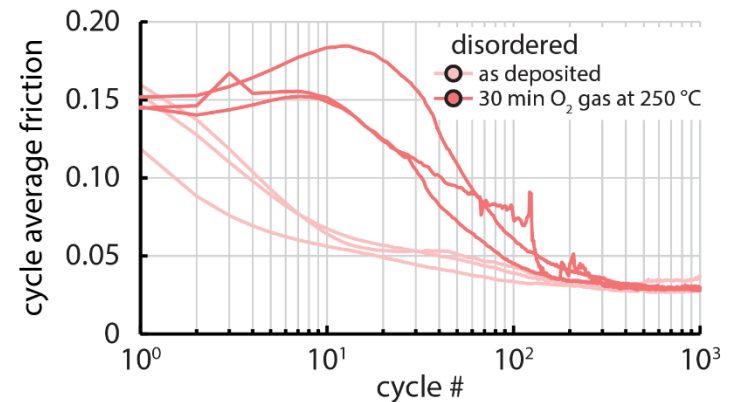
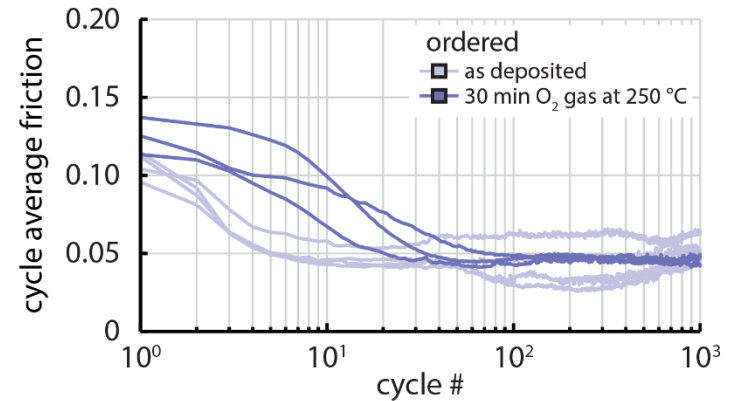
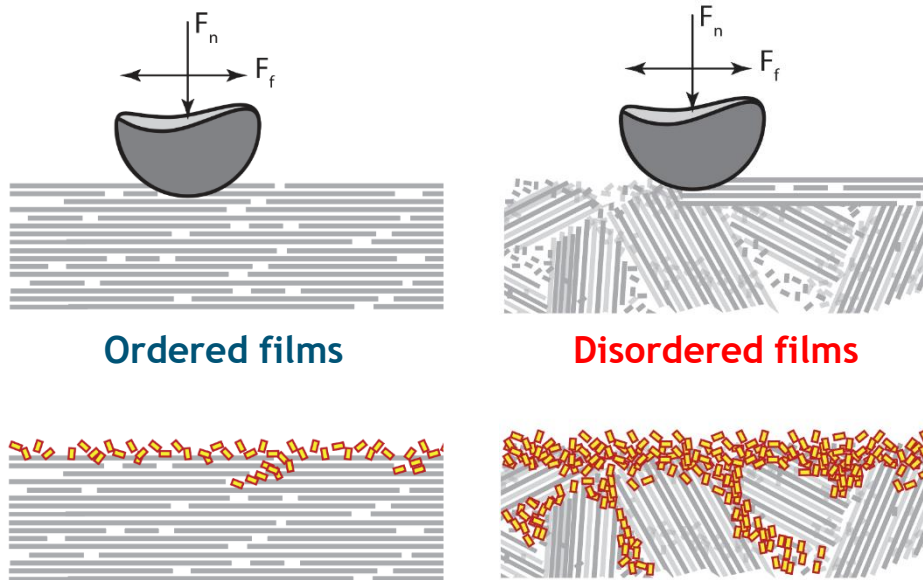
highly ordered (N₂ sprayed) [1]



- To make things simple, we focused on ordered, impinged films
 - Blast N₂/MoS₂ onto surface to get films close to basally oriented as deposited
 - Exhibit lower initial friction coefficients
 - Ordering may help prevent degradation

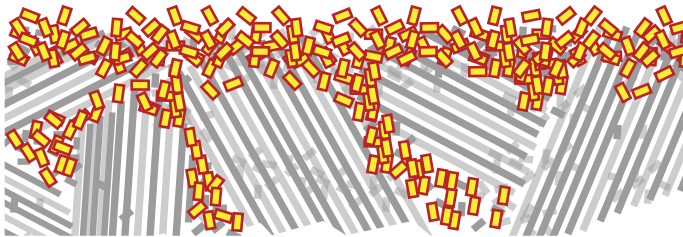


Accelerated Aging Study

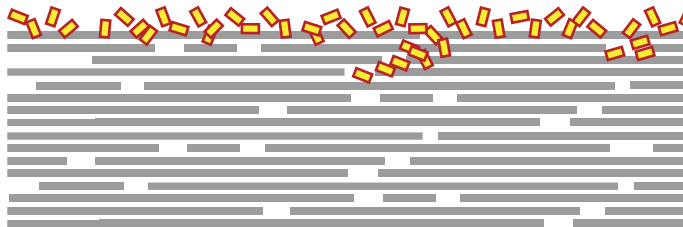


- Less defects/edges may also limit oxidation to surface
- Exposures of 30 min O₂ @ 250°C (also AO) show ordered films exhibit less oxidation & surface limited, reducing effects on run-in

First demonstration of microstructure's role in limiting oxidation; run-in (ordered) surfaces help prevent aging related issues



disordered structure



highly-ordered structure

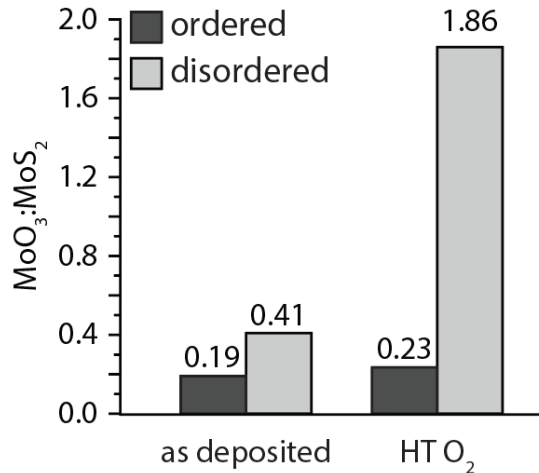
- Oxidation resistance should benefit in the same way that run-in does from ordered surfaces

- Higher degree of basal orientation and less available edge site (large crystals) should reduce oxidation

- Ordered structure also provides more tortuous path into the bulk for further interactions



Mo 3p signal - MoO₃:MoS₂ ratio

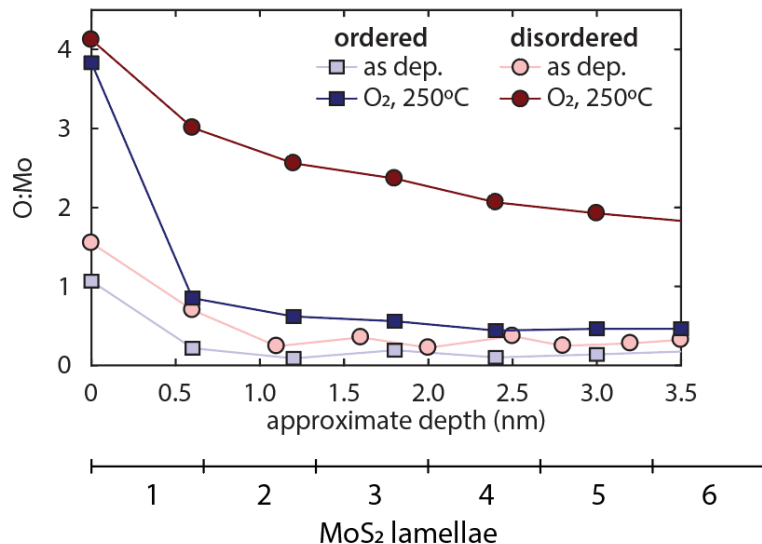


- Look at amount of Mo as sulfide or oxide after exposures to O₂ @ 250°C and Atomic Oxygen (30 min)

Oxygen Gas (30 min @ 250°C)

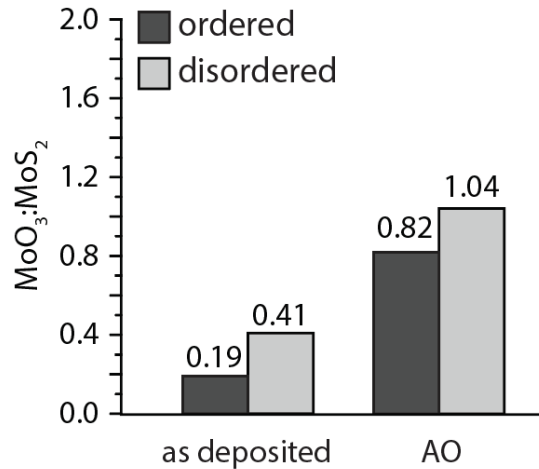
- XPS indicates minimally more oxide for ordered films while disordered films have more
- LEIS shows this is mostly surface limited for ordered films and through the surface for disordered

oxygen : molybdenum ratio





Mo 3p signal - MoO₃:MoS₂ ratio



- Look at amount of Mo as sulfide or oxide after exposures to O₂ @ 250°C and Atomic Oxygen (30 min)

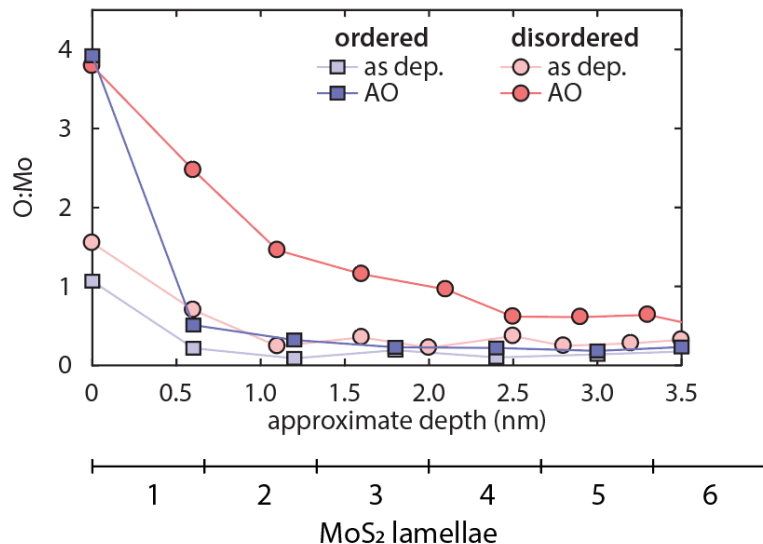
Oxygen Gas (30 min @ 250°C)

- XPS indicates minimally more oxide for ordered films while disordered films have more
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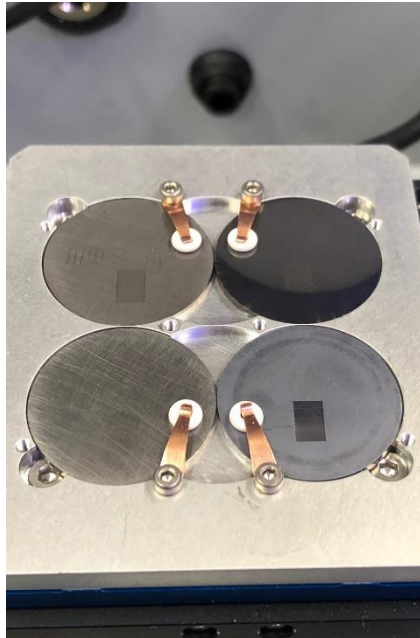
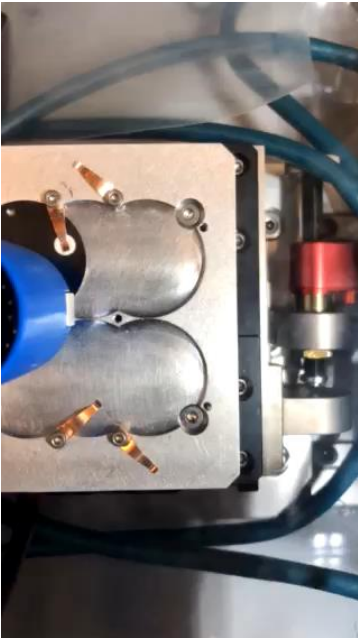
Atomic Oxygen (30 min @ RT)

- AO exposures show similar increases in oxidation via XPS
- Again LEIS shows oxygen only at surface for ordered films and not much below the surface for disordered

oxygen : molybdenum ratio



- Perform XPS inside versus outside rubbed area
- Return to run-in area after aging for additional friction testing



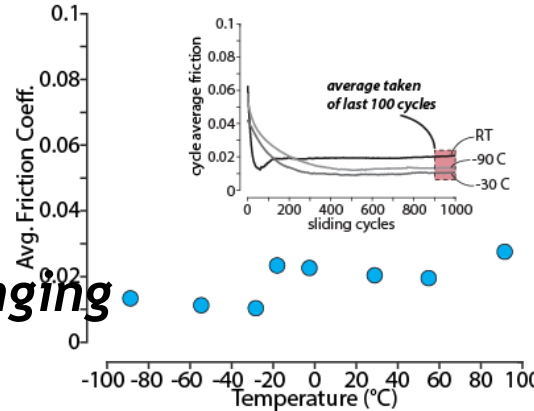
Link #4: Structurally driven temperature dependent friction of MoS₂ - friction



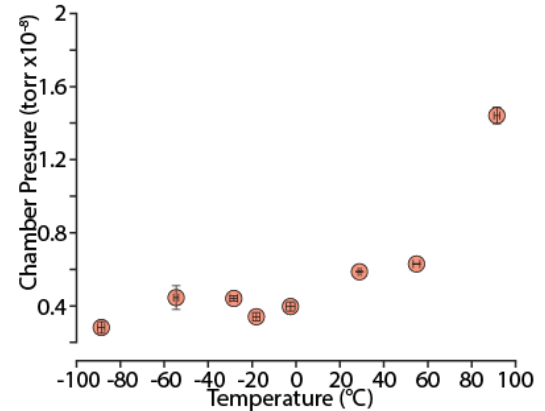
Constant Pressure, Changing Temperature

- In UHV, at constant pressure, the friction behavior of MoS₂ is INDEPENDENT of temperature
- Coating shows no measurable wear at any temperature

A) Temperature Independent Friction



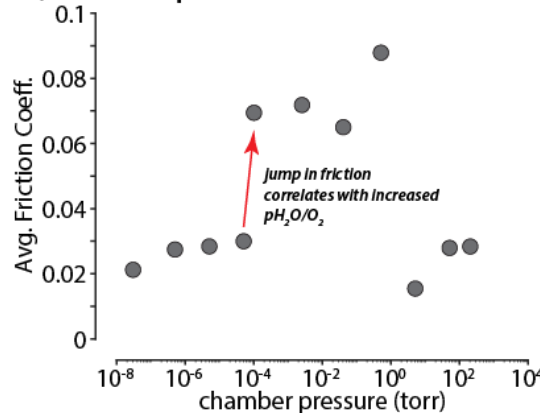
B) Chamber Pressure at Various Temperatures



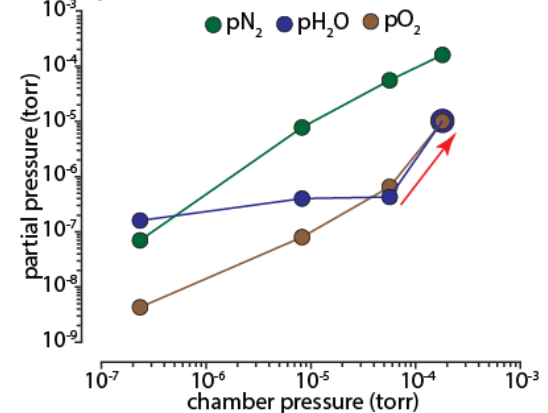
Constant Temperature, Changing Pressure

- In UHV, at constant temperature, the friction behavior of MoS₂ changes with pressures
- A high friction regime at moderate vacuum ranges
- Friction drops again at higher pressures
- The increase in friction correlates with an increase in the pH₂O and pO₂

C) Pressure Dependent Friction



D) Partial Pressures from RGA



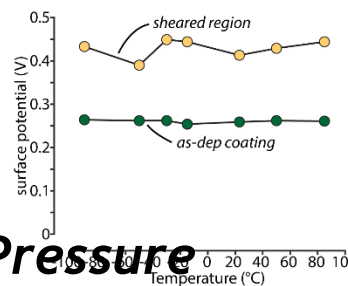
Question: How is structure driven by temperature and pressure?



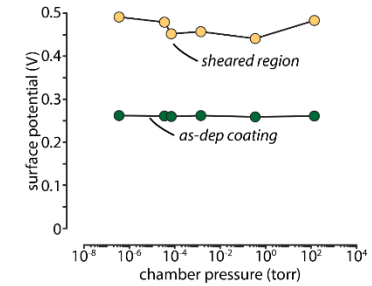
Constant Pressure, Changing Temperature

- Sheared regions at all temperatures have higher surface potential than coating
- No apparent trend between friction and surface potential
- **Constant Temperature, Changing Pressure**
- ~~potential~~ increase in surface potential = increase in friction coefficient
- Increase in surface potential = decrease in friction coefficient
- Strong relationship between friction and surface potential

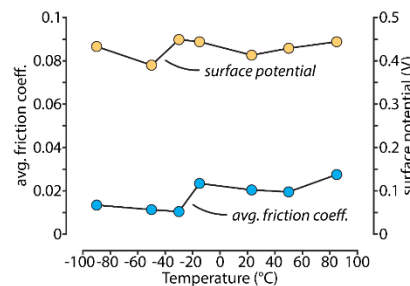
A) Surface Potential vs. Temp



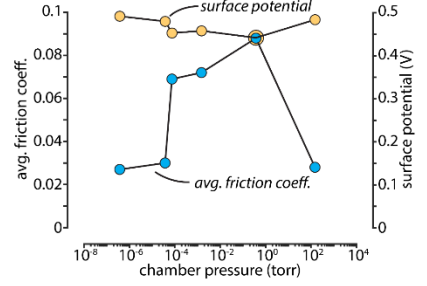
D) Surface Potential vs. Pressure



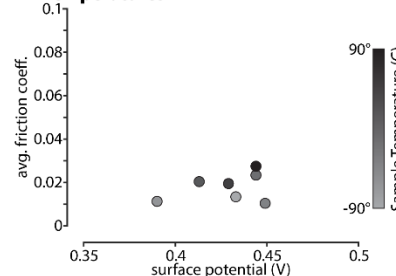
B) Surface Potential and Friction vs. Temp



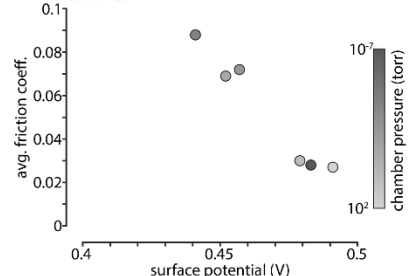
E) Surface Potential and Friction vs. Pressure



C) Surface Potential vs. Friction at different Temperatures



F) Surface Potential vs. Friction at different Pressures



Takeaway: Structure does not evolve with temperature, only pressure