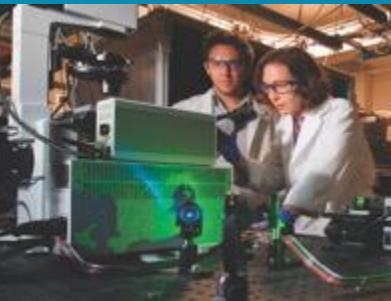


# Structural, Environmental and Mechanochemical Interactions in Solid Lubricant Material Systems



**J.F. Curry**

Sandia National Laboratories, Albuquerque, NM

**JAST Webinar**

**8 December 2021**

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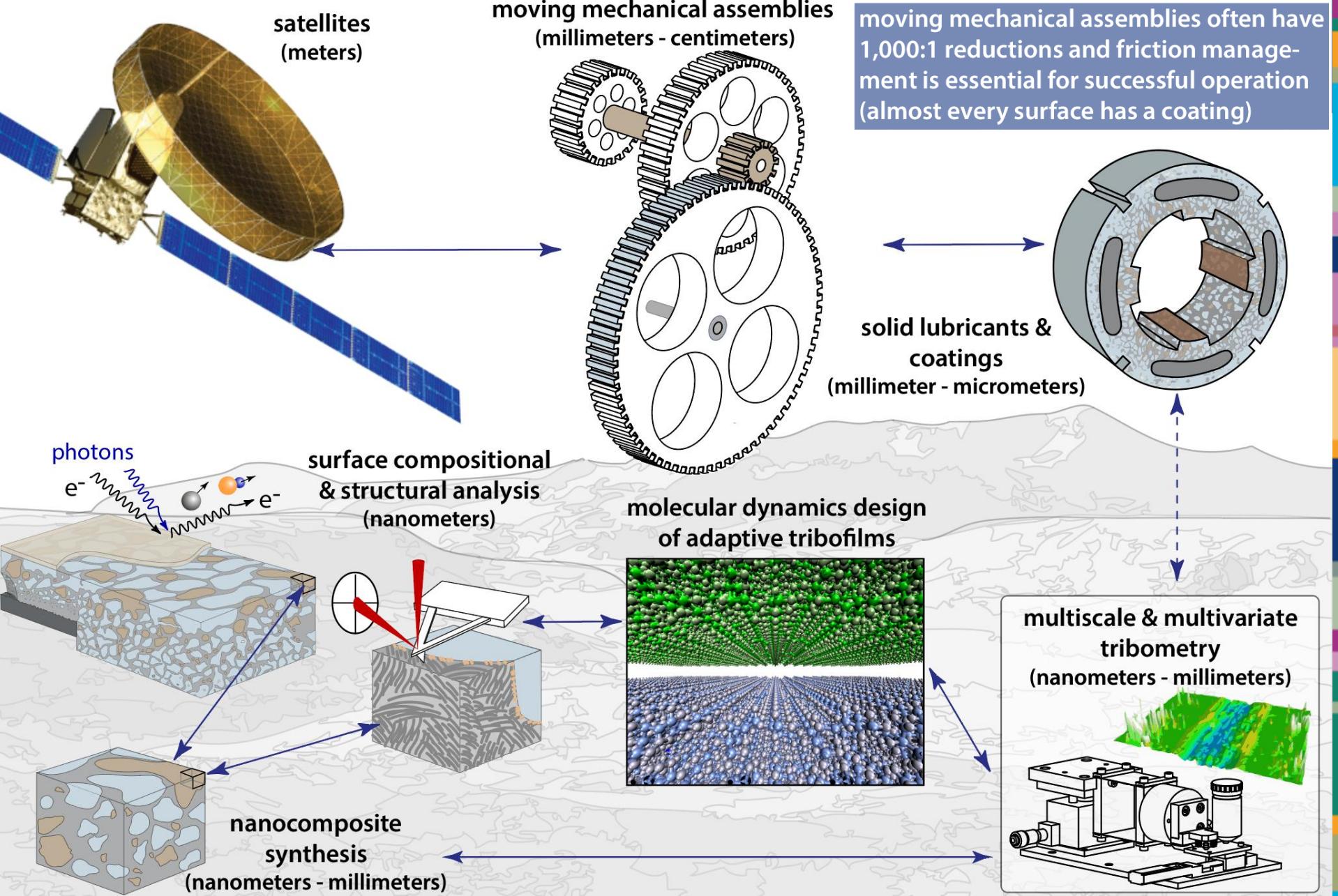


## 1. Brief Introduction & Motivation

## 2. Current & Ongoing Research Projects

- Environmental Factors Governing Run-In Behaviors and Oxidation response in MoS<sub>2</sub>
  - Aging of Pure/Composite films with Varying Structures
  - ALD MoS<sub>2</sub> Conversion
  - Fundamental Role of Structure on Friction Response
- Understanding Variability in Run-in for DLC Coatings
  - High Throughput Assessment of Different Film Compositions
  - Deposition Directed Surface Termination Studies
- Self-lubricating *in situ* carbon films (SLIC)

# Fundamental Studies & Applied Challenges



# Extreme Operating Environments



## Space:

- operate in vacuum (atomic oxygen in low earth orbit)
- store months – years before use; generally non-serviceable
- operating temperatures from 50 – 300K, depending on location
- large investments of time and money



# Extreme Operating Environments



## Space:

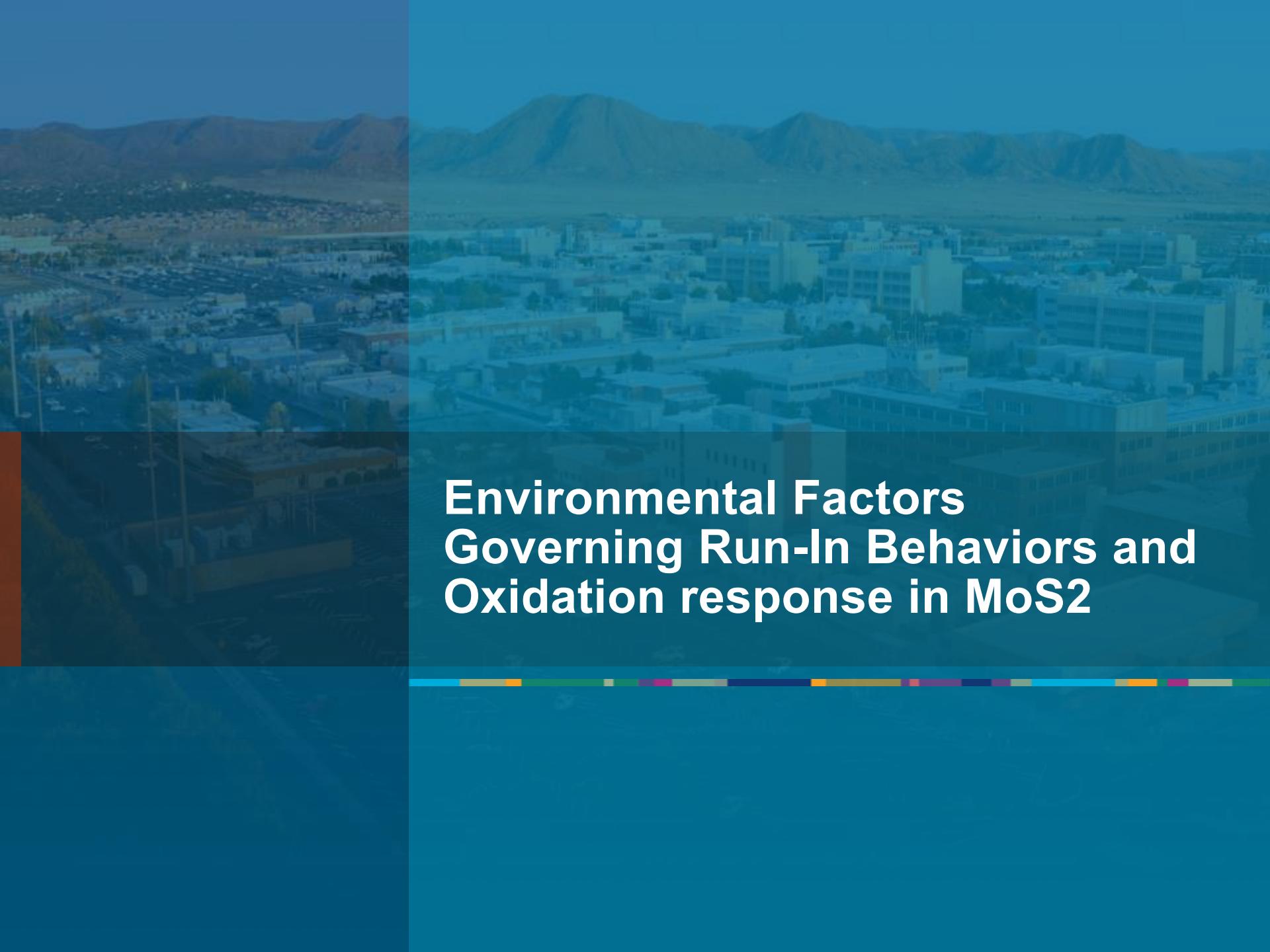
- operate in vacuum (atomic oxygen in low earth orbit)
- store months – years before use; generally non-serviceable
- operating temperatures from 50 – 300K, depending on location
- large investments of time and money



## Precision Mechanisms:

- inert gas near  $P_{atm}$ , trace  $O_2$ ,  $H_2O$ , outgassing species
- store for decades; non-serviceable
- operating temperatures 200 – 350K
- large investments of time and money
- consequences (political, societal) of failure are unacceptable





# Environmental Factors Governing Run-In Behaviors and Oxidation response in MoS<sub>2</sub>

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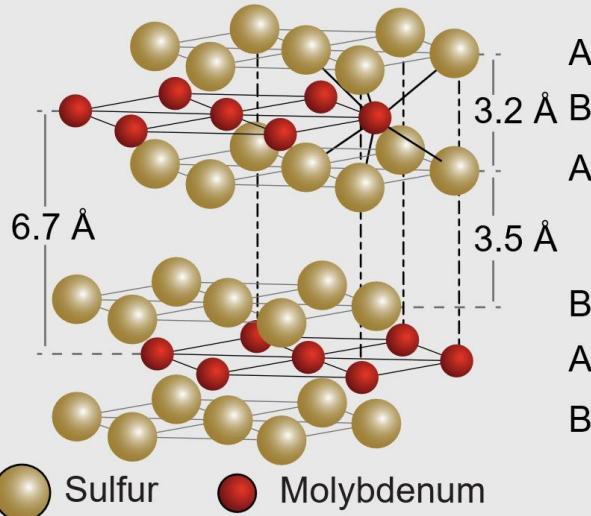
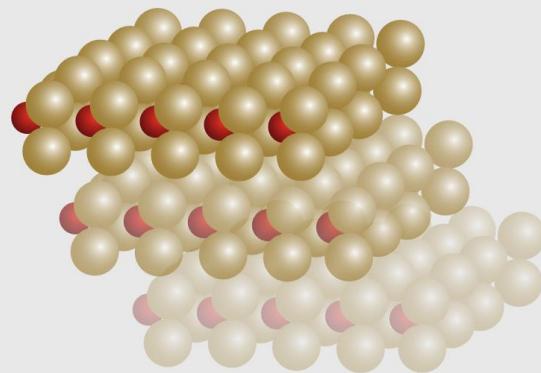
# MoS<sub>2</sub> – How it Works



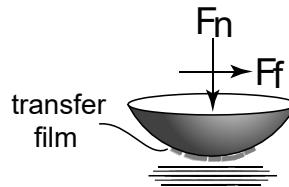
molybdenum disulphide

$\mu = 0.02 - 0.06$  (inert @ 1N)

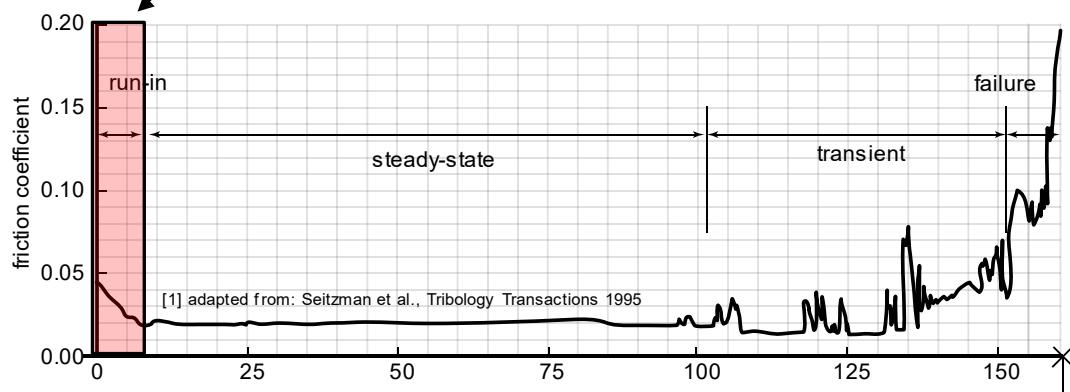
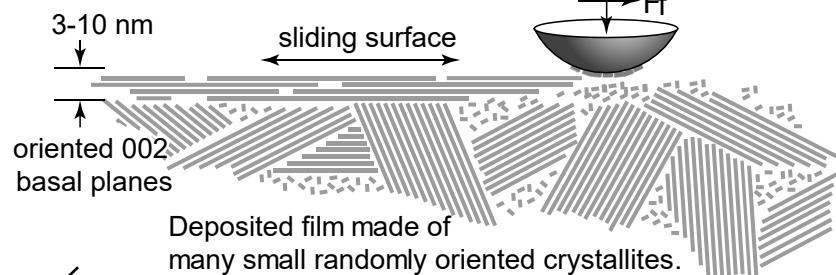
$\mu = 0.15 - 0.25$  (humid air @ 1N)



generalized run-in processes

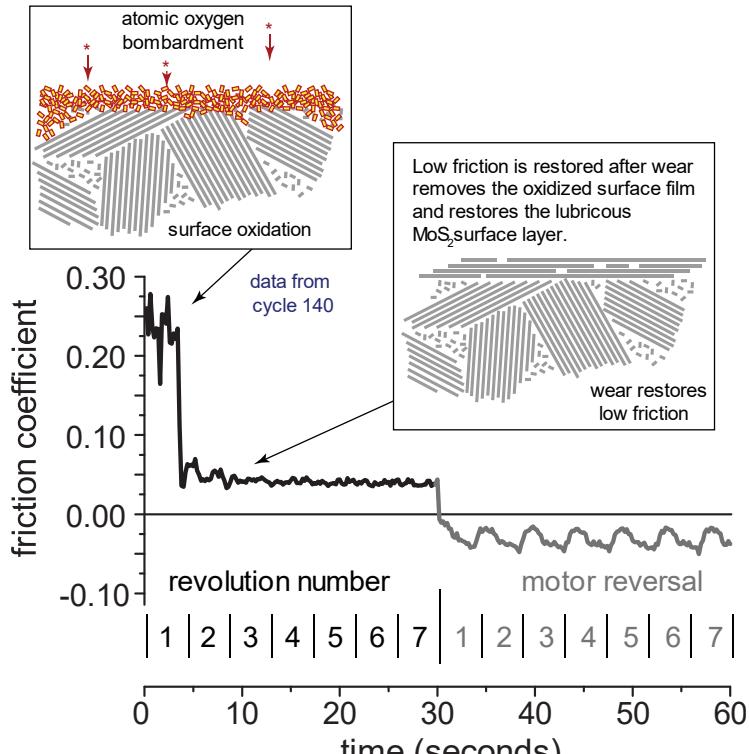


2) Shear-induced  
crystallite re-orientation



Typical friction trace for MoS<sub>2</sub> lubricated contact. Initial run-in is followed by steady state low friction which ultimately transitions to high friction before failure [1]

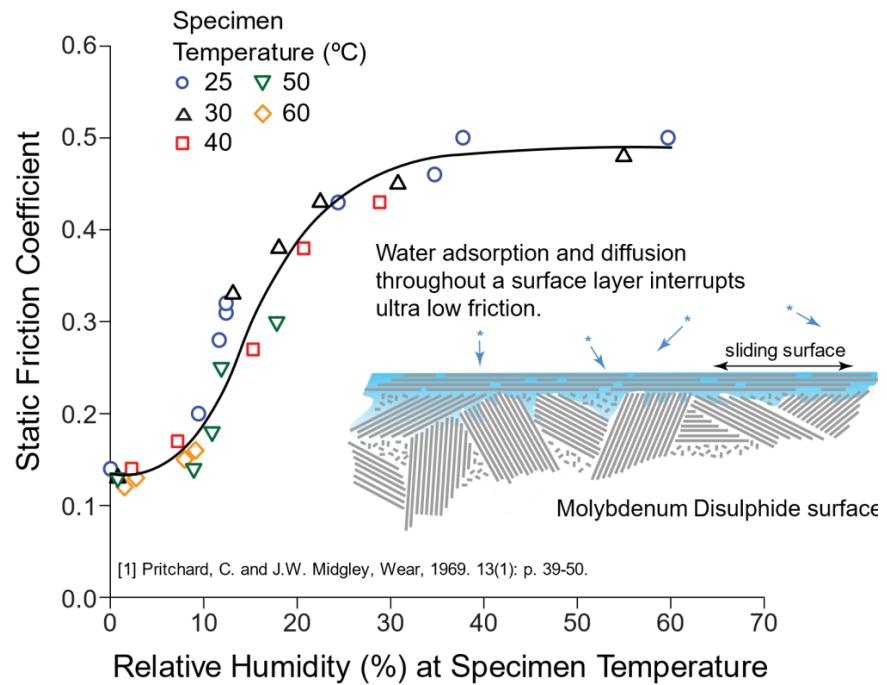
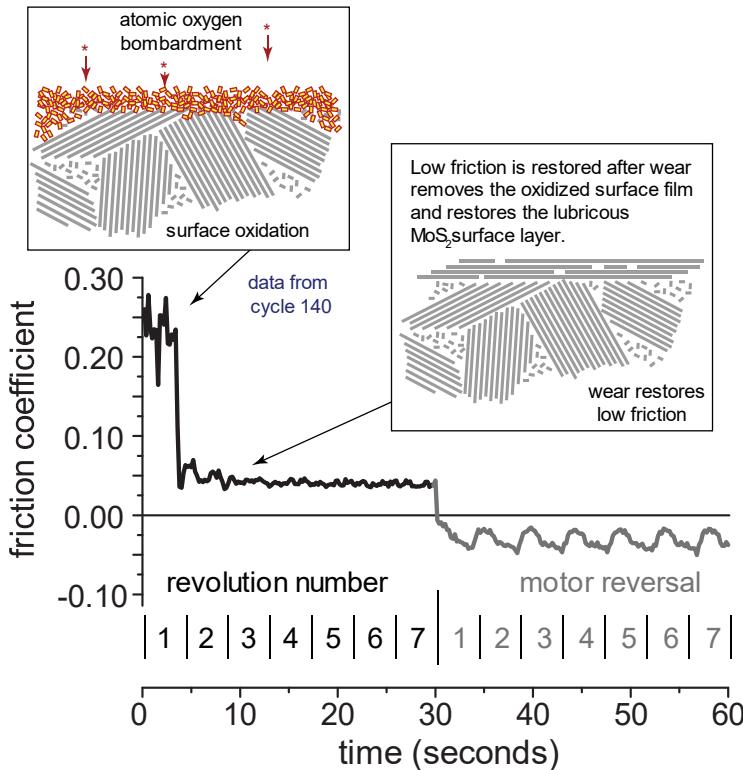
# Bad Actors – Environment & Aging



Krick et. al, unpublished

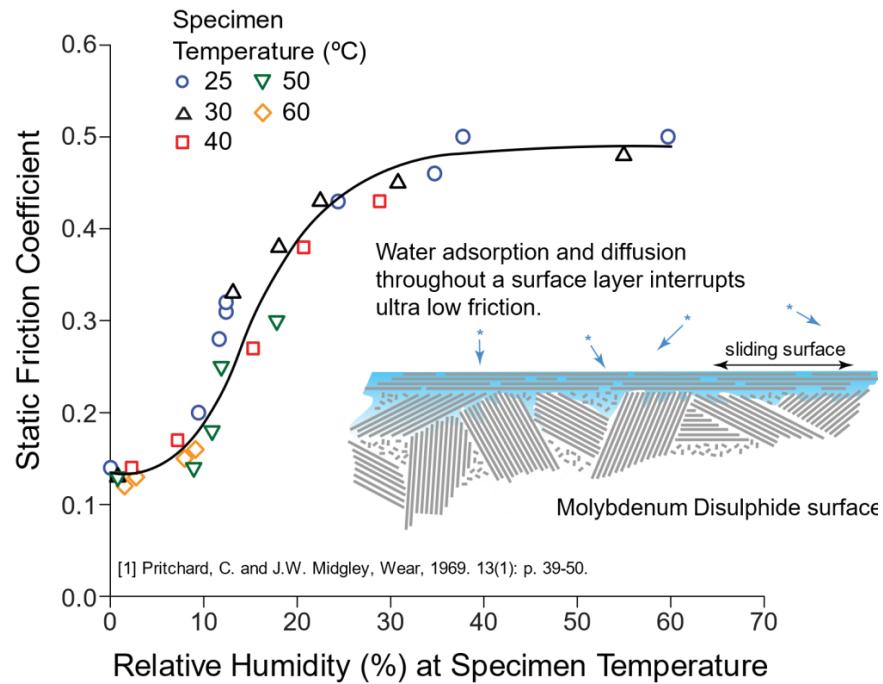
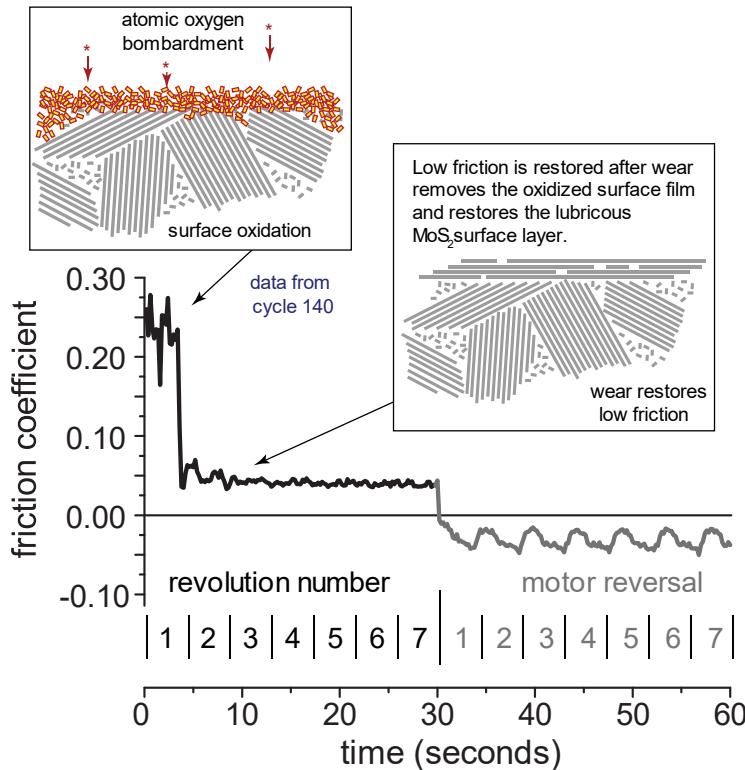
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Many components operate infrequently and for very few cycles – **effectively living in the run-in regime**

# Initial Friction Evolution (Run-In)

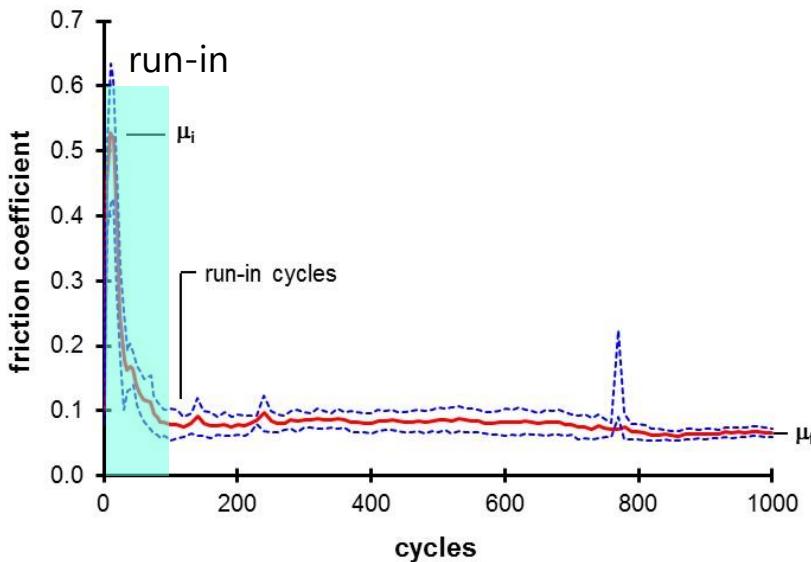


## Drawbacks

- Increased friction/wear
- Risk of seizure
- Reduced performance margin in mechanisms
- Loss of energy

## Causes

- Structural (re-orientation, crystallization)
- Environmental (oxidation, adsorbates)



## Solutions

- Tailor composition (compositing for densification, water-getting)
- Tailor microstructure, encourage ordering (impingement films, capping layers, deposition inputs)

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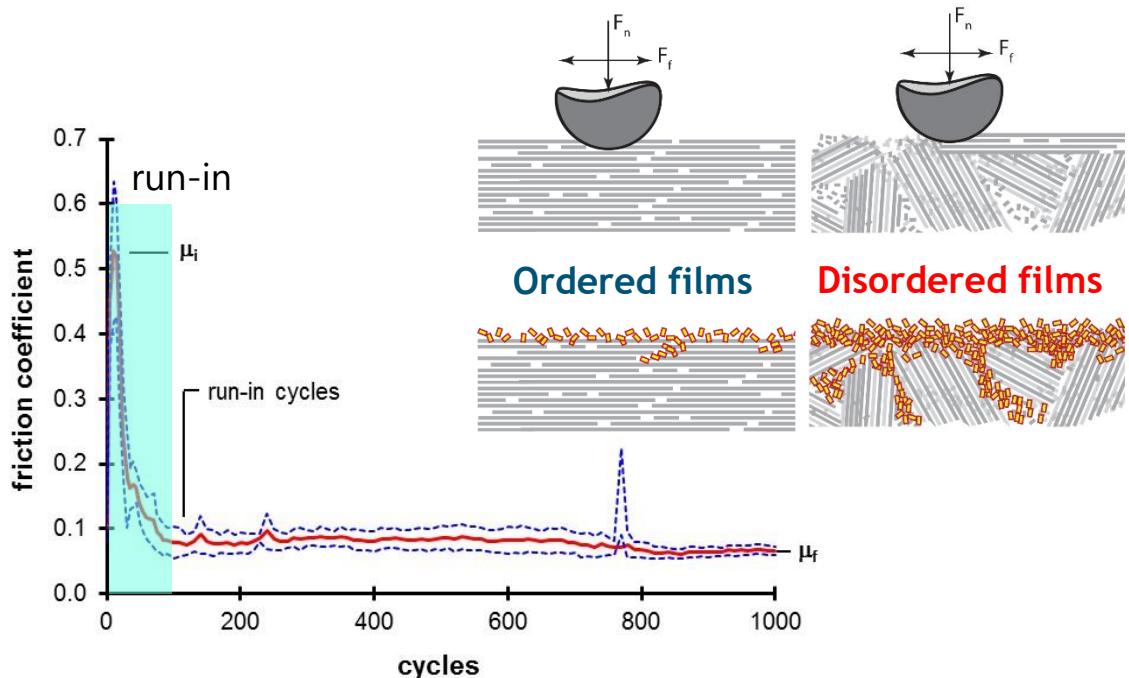
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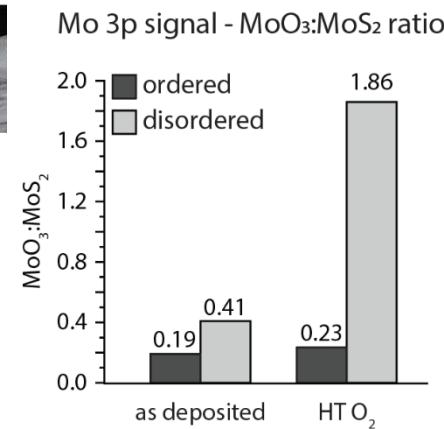
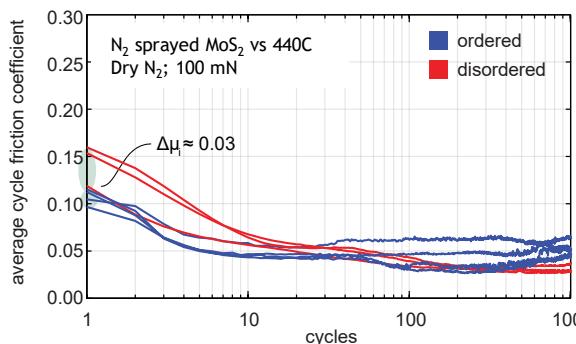
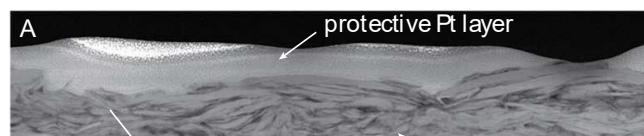
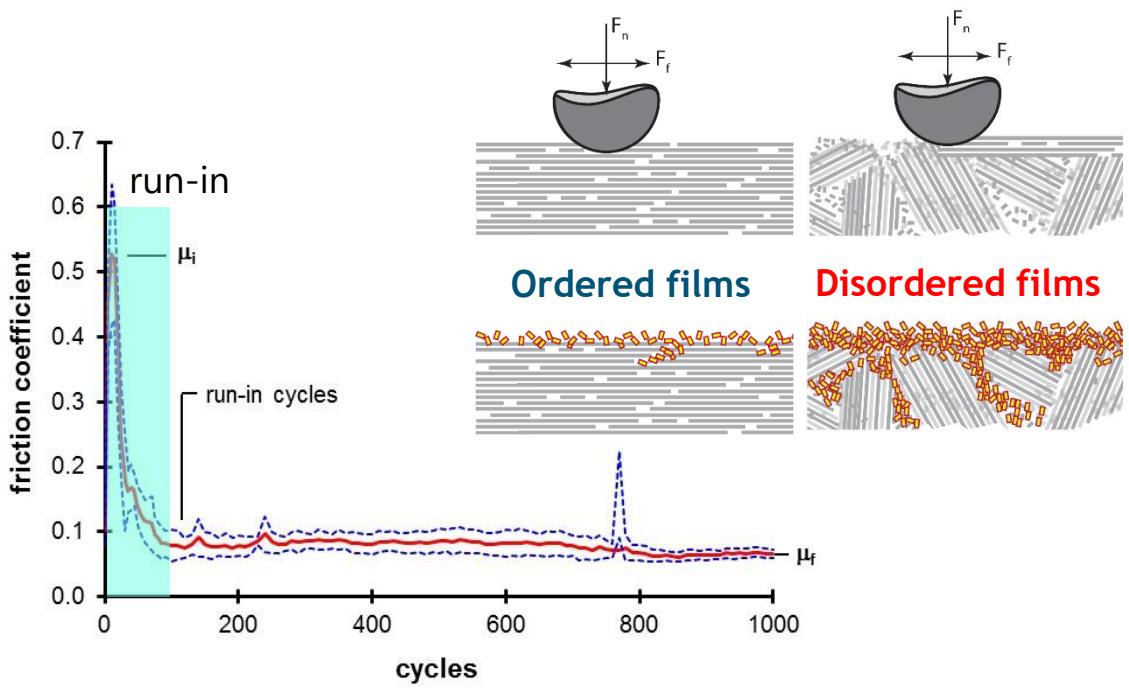
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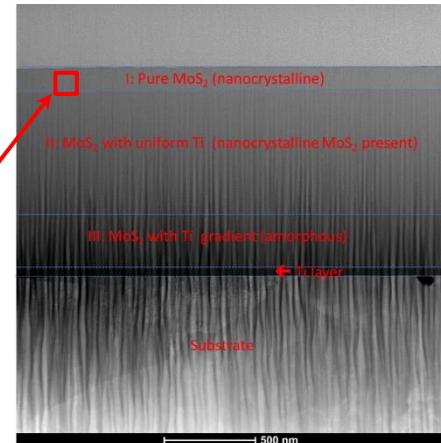
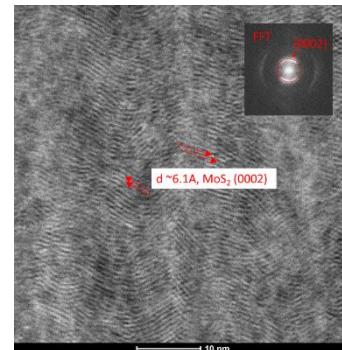
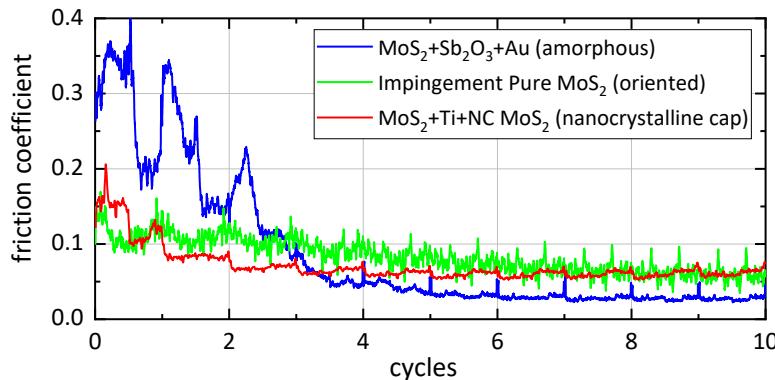


# Efforts to Reduce Run-In: Structure/Composition

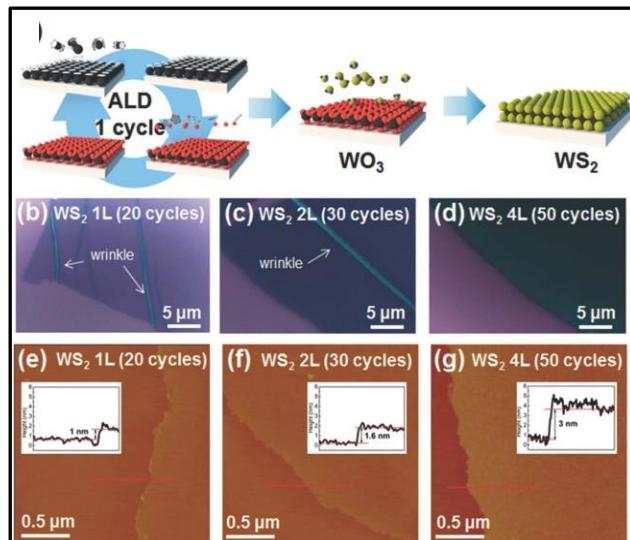


## PVD nanocrystalline & composite films

- Films with NC structures exhibit reduced run-in compared to amorphous films (composites); likely reduced reactivity as well



MoS<sub>2</sub>+Ti+pure MoS<sub>2</sub> cap



## MoS<sub>2</sub> ALD at LBNL Molecular Foundry

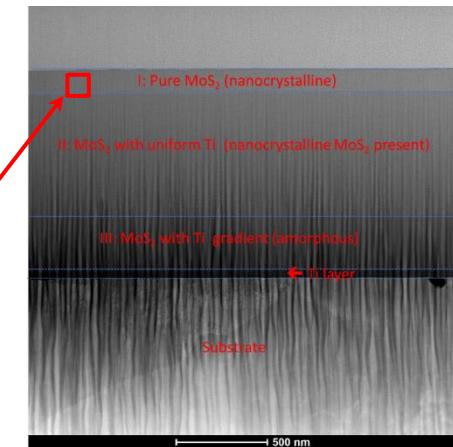
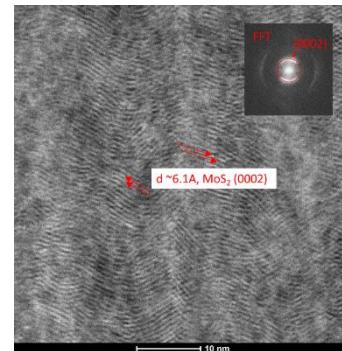
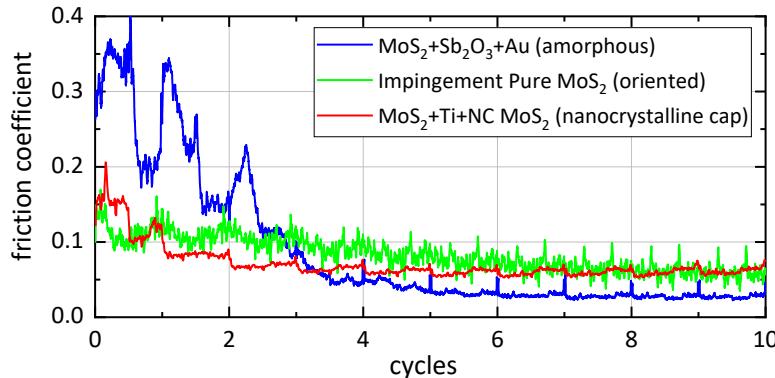
- Developing ALD MoS<sub>2</sub> freestanding and capped films for aging resistance
- MoO<sub>x</sub> conversion with Moly hexacarbonyl precursor and H<sub>2</sub>S conversion
- Challenges remain in optimizing synthesis to promote full conversion at low enough temperatures

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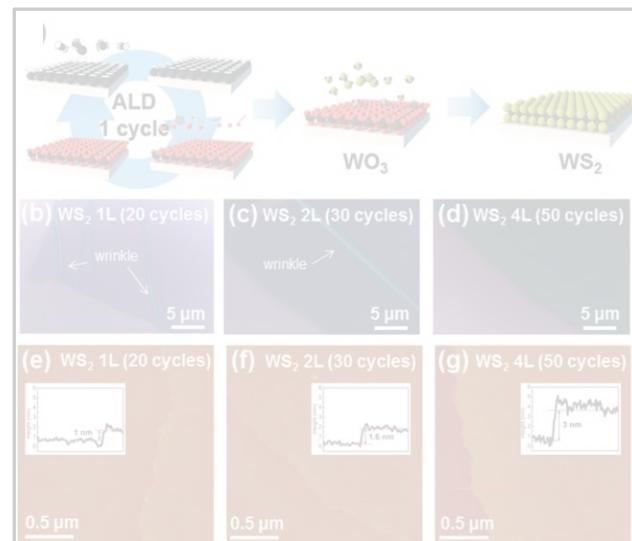


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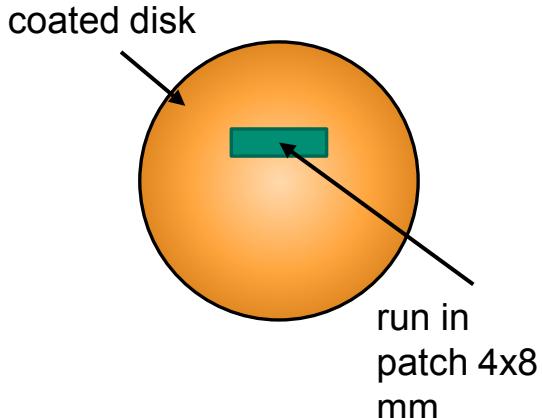
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# Aging Study - Experimental Methods



## Run In



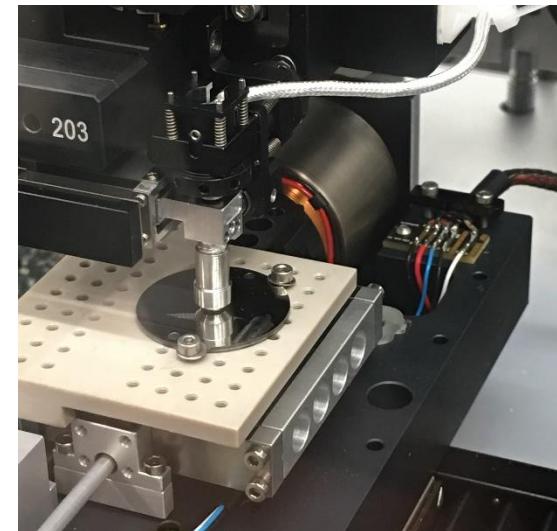
- 13-8PH or 440C stainless steel disks
- run in at 530 MPa, 50 passes, overlapping areas

## Accelerated Aging



- 200°C, dry (DP < -60°C) air, 5 SCFH
- 12 hours

## Friction Testing



- 440C ball, 3.2 mm dia.
- 1 mm/s sliding speed
- Hertz contact pressures of 275, 530 and 785 MPa

## Materials Investigated:

- N<sub>2</sub> (pure MoS<sub>2</sub> sprayed with N<sub>2</sub>)
- DC (pure DC sputtered MoS<sub>2</sub>)
- Ti (RF sputtered MoS<sub>2</sub>, Ti-doped)
- Sb<sub>2</sub>O<sub>3</sub>/Au (RF sputtered Sb<sub>2</sub>O<sub>3</sub>+Au-doped MoS<sub>2</sub>)

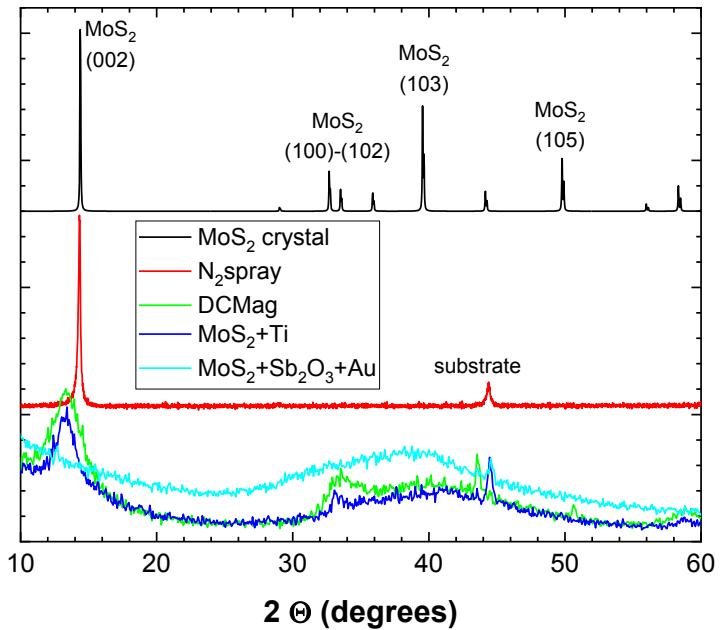
Pure MoS<sub>2</sub>

Doped MoS<sub>2</sub>

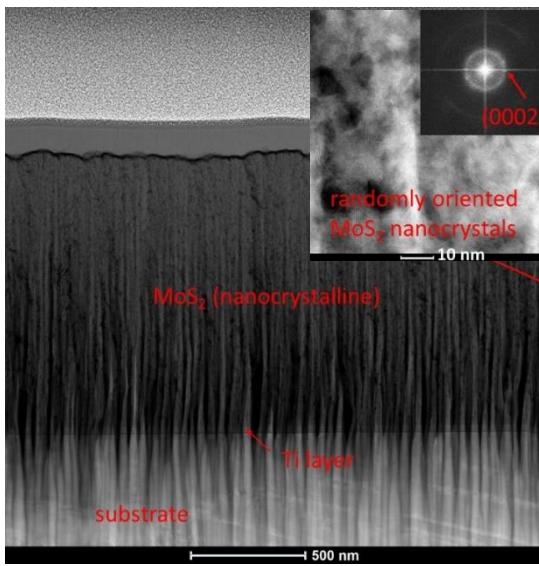
# Coating Microstructures & Compositions



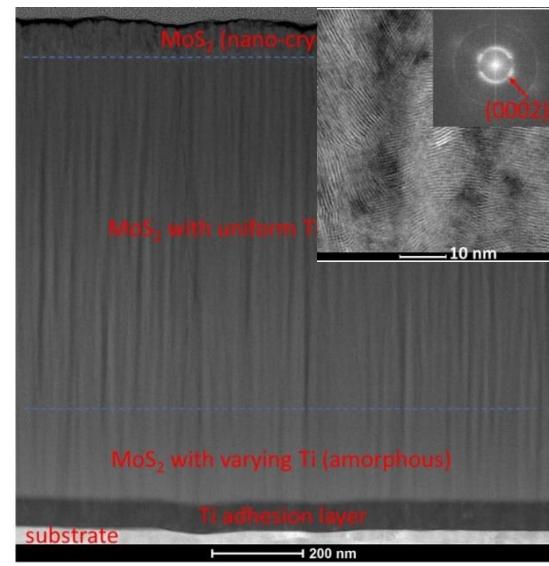
Intensity (a.u.)



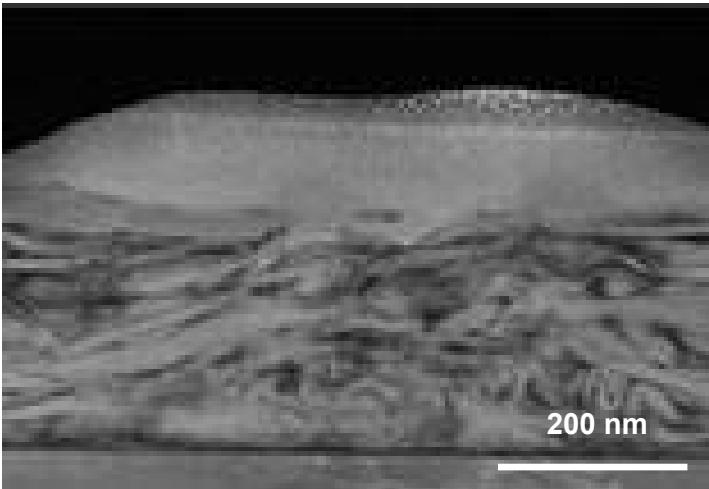
DC Mag pure MoS2



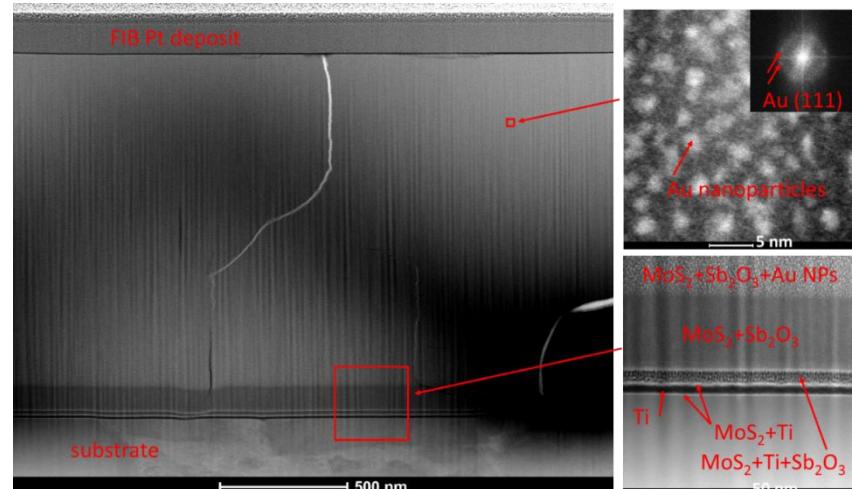
MoS2/Ti + pure MoS2



N2 spray pure MoS2



MoS2/Sb2O3/Au

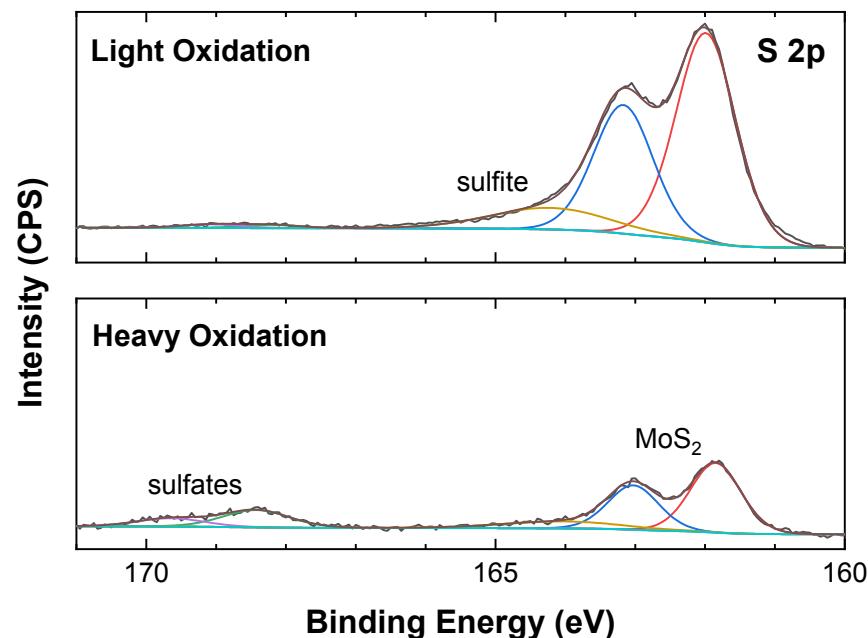
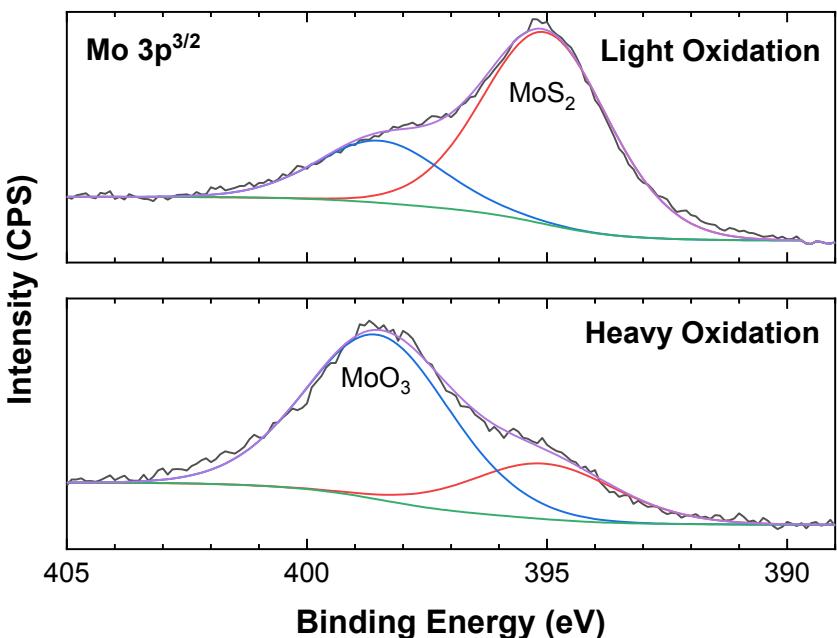


# Quantifying MoS<sub>2</sub> Oxidation via XPS

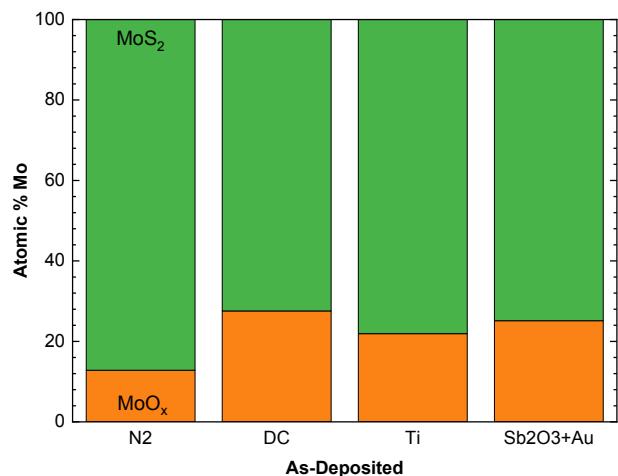
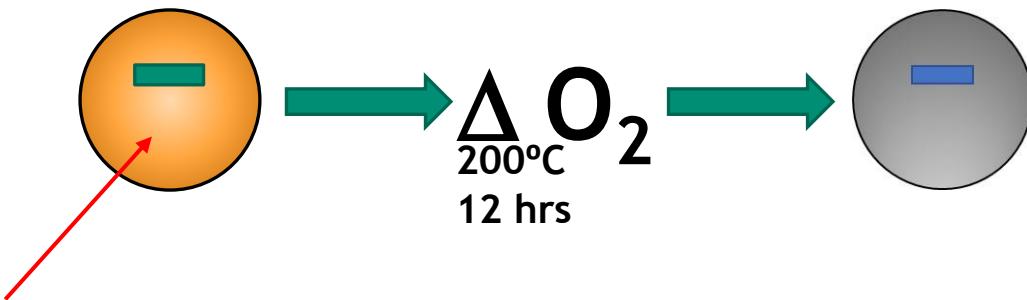


X-ray Photoelectron Spectroscopy (XPS) for surface chemical analysis

- survey scan for concentration of major elements present
- detailed scans of Mo3p, S2p spectral regions
- deconvolution of detailed scans to determine amount of Mo, S bonded to one another compared to oxidized species (MoO<sub>3</sub>, sulfates, sulfites, etc.)
- surface sensitive – analyzing the top few nanometers

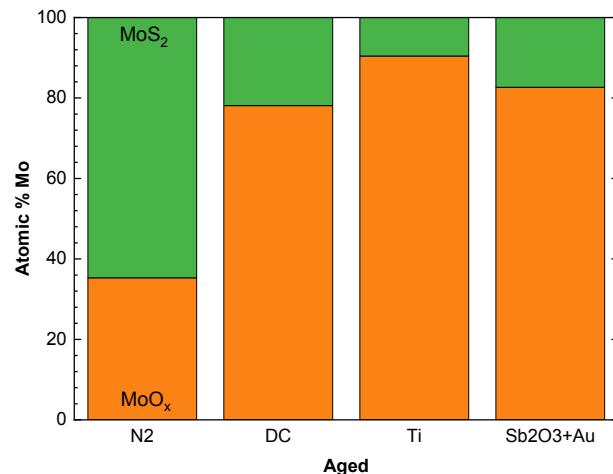
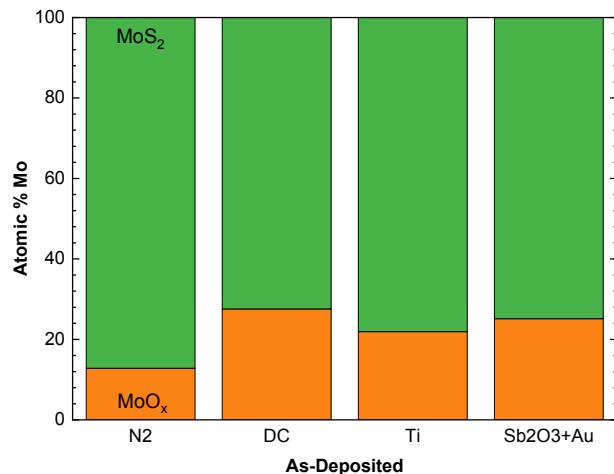
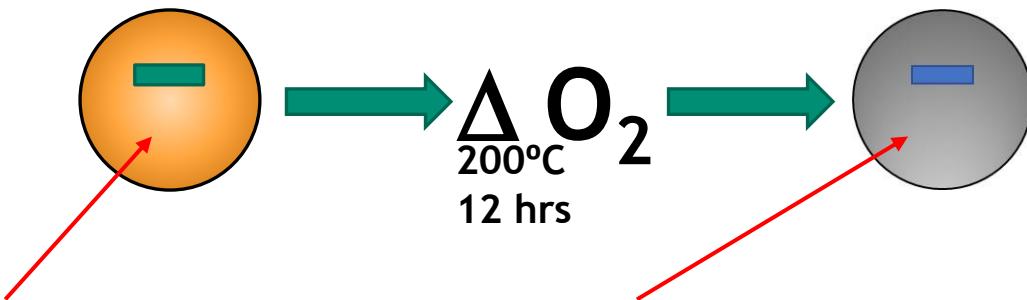


# XPS Results – Pure



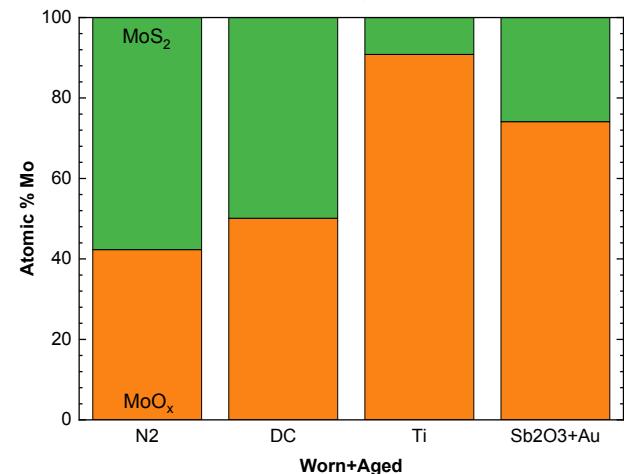
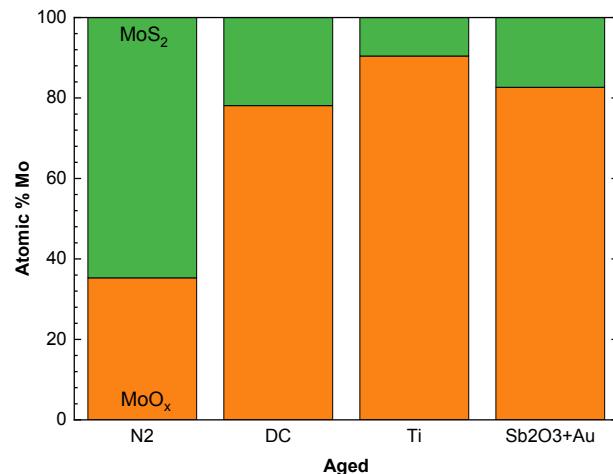
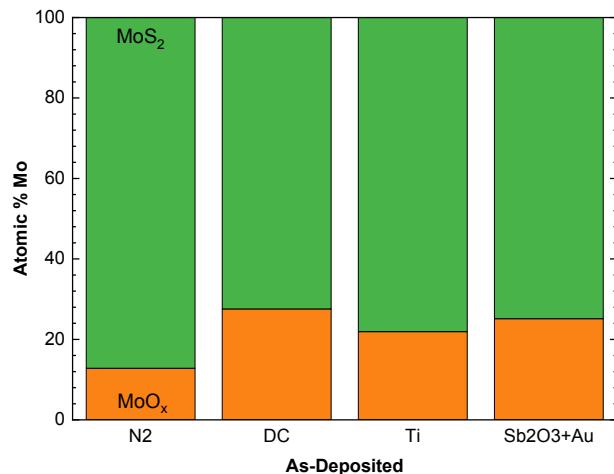
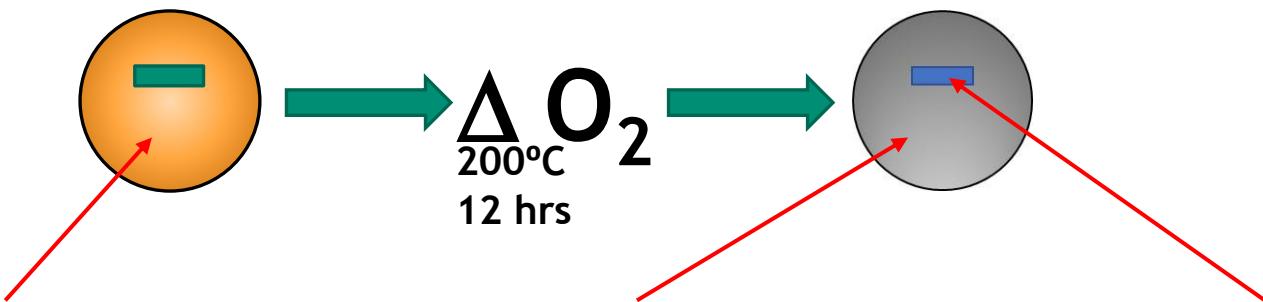
- Sprayed films natively contain less oxide than what is taken up during sputtering

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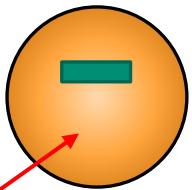
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- Sprayed films in unworn state also less susceptible to oxidation

# XPS Results – Pure

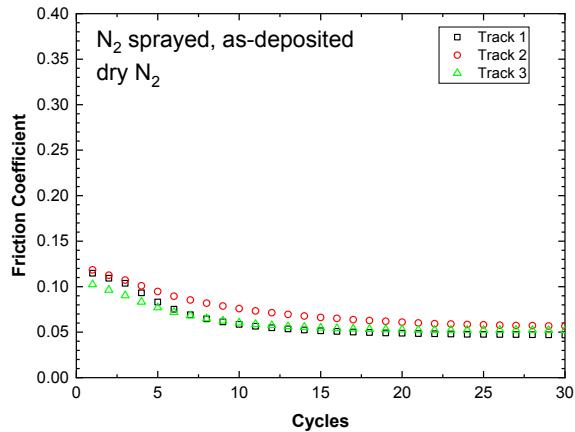


- Sprayed films natively contain less oxide than what is taken up during sputtering
- Sprayed films in unworn state also less susceptible to oxidation
- Worn area for DC mag films behaves like sprayed films, preventing oxidation
- Composite films run-in do not appear to buy significant protections to oxidation as compared to pure counterparts

# Friction Results – Pure MoS<sub>2</sub> Films

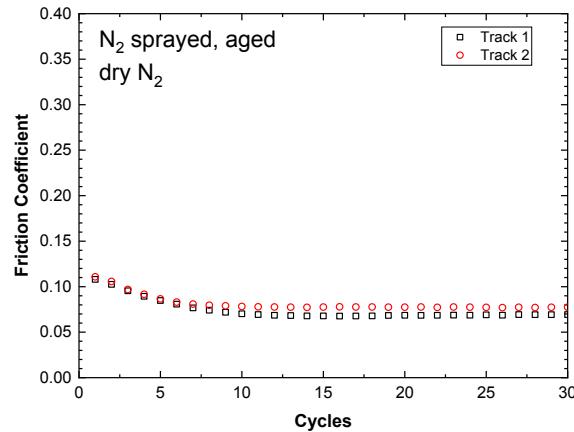


As Deposited

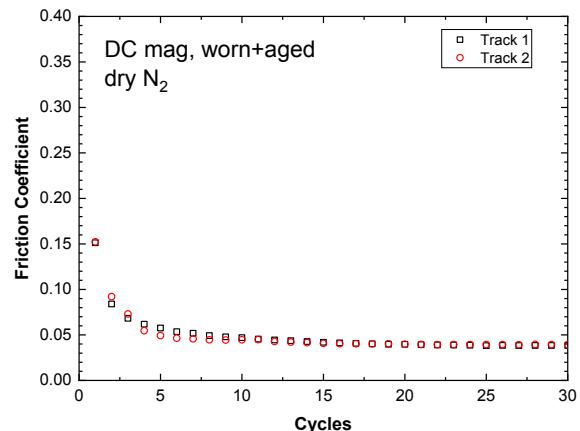
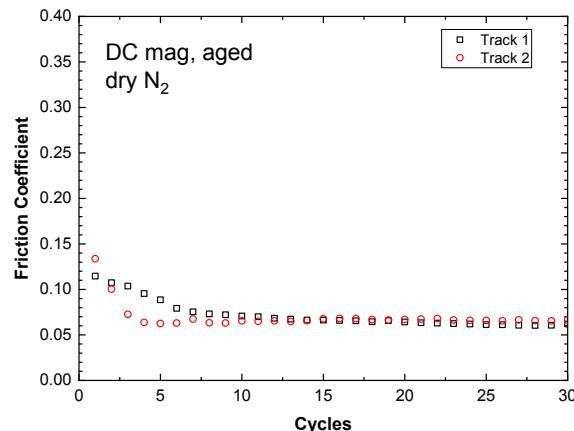
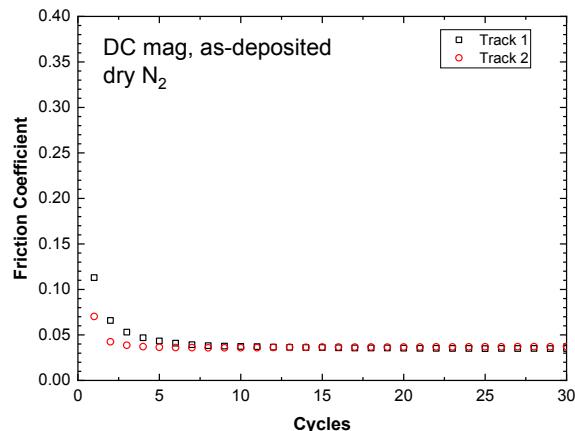
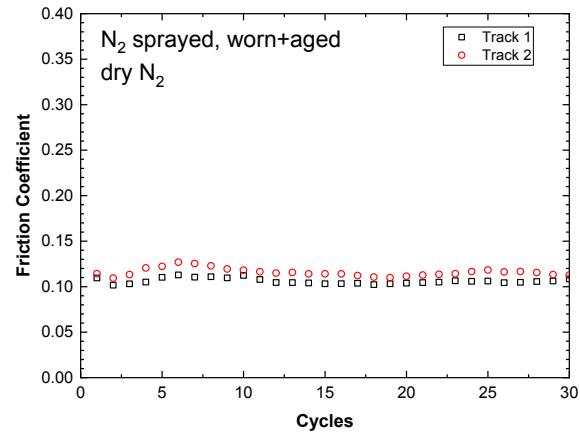


$\Delta O_2$   
200°C  
12 hrs

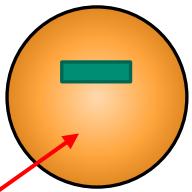
Aged



Worn + Aged

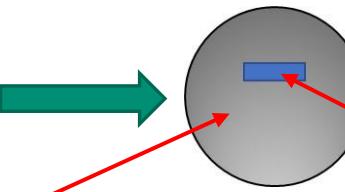


# Friction Results – Composite MoS<sub>2</sub> Films



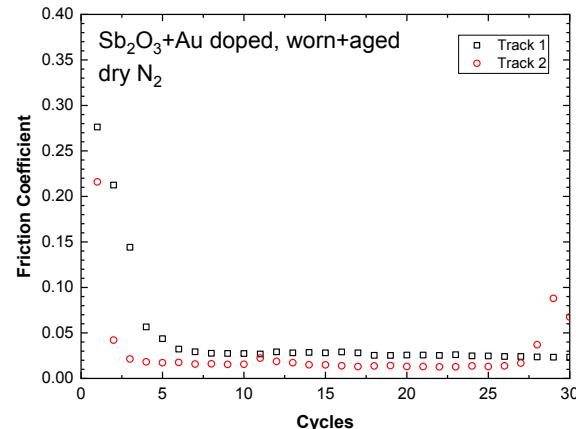
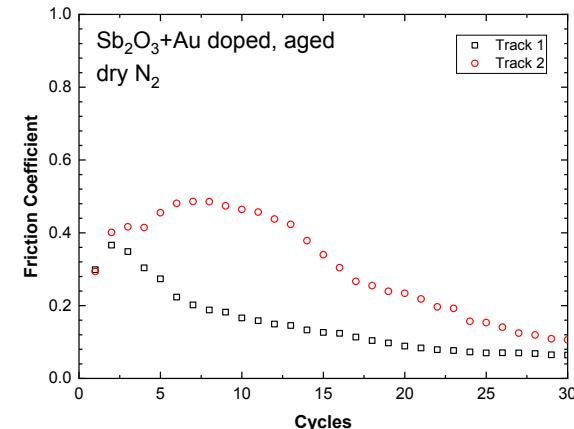
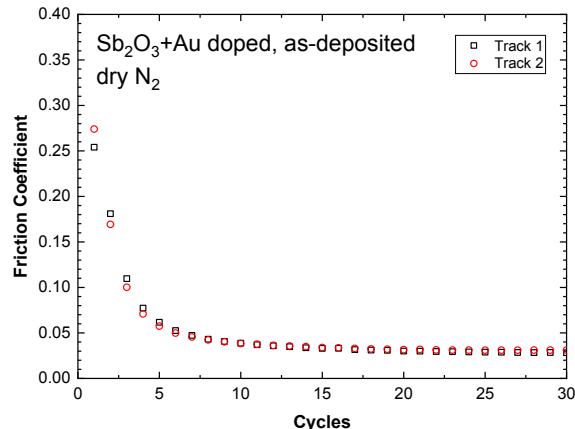
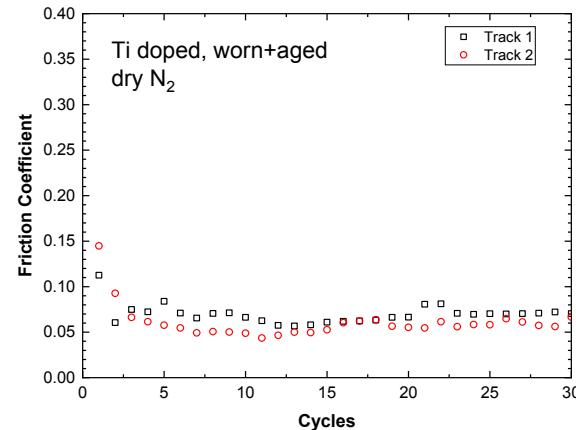
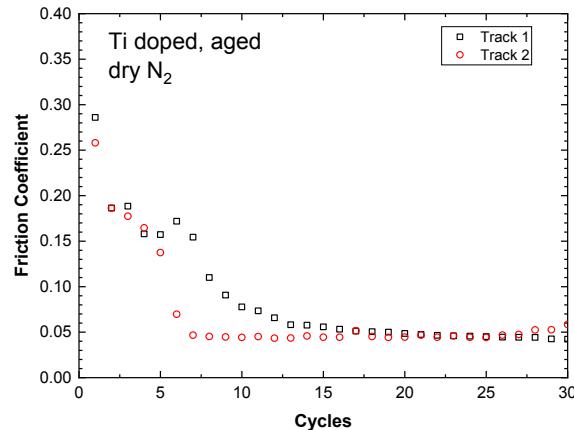
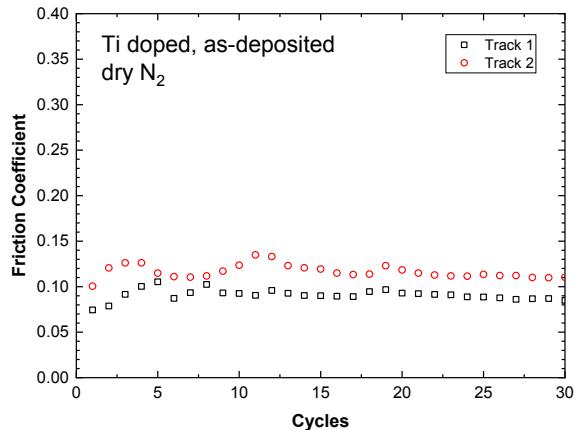
As Deposited

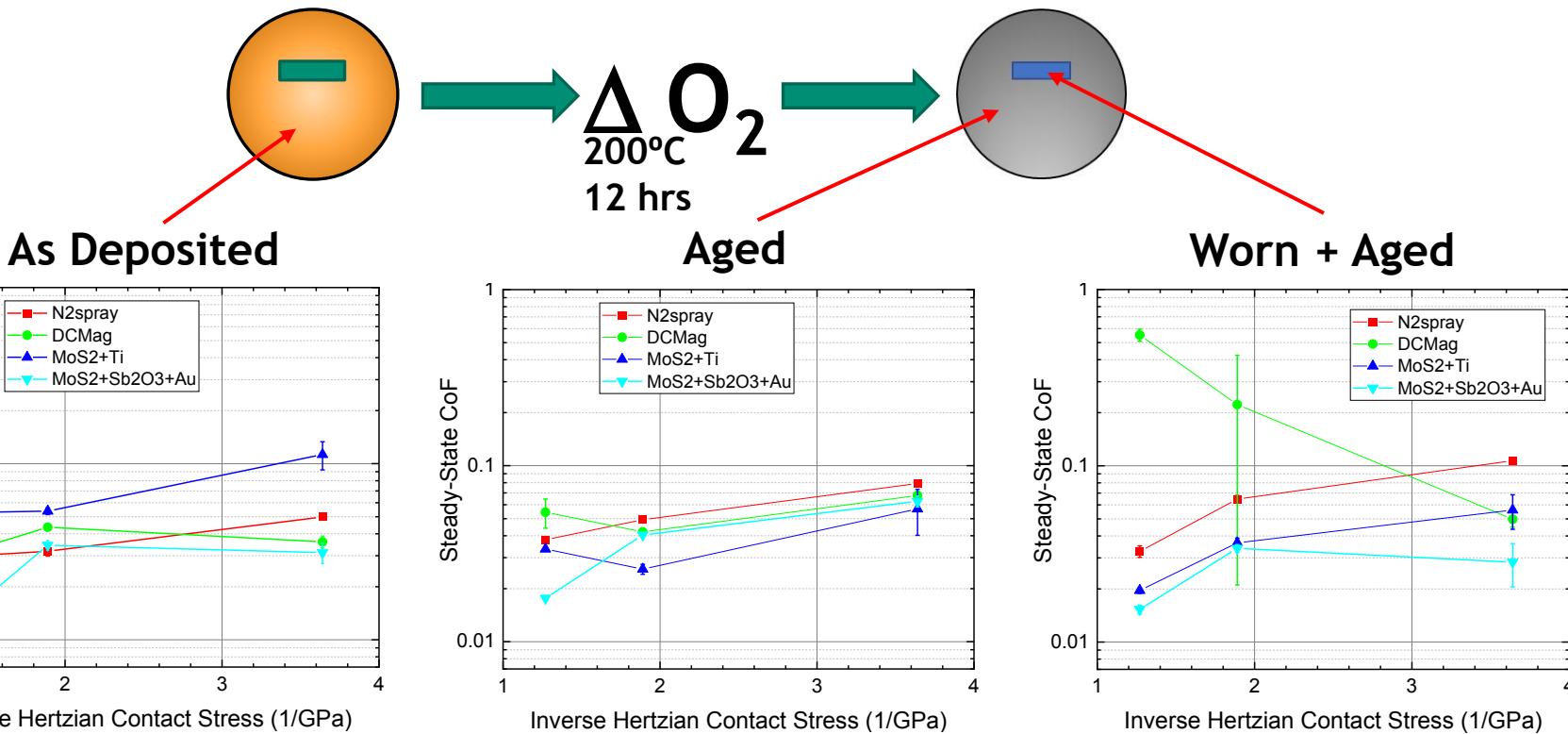
$\Delta O_2$   
200°C  
12 hrs



Aged

Worn + Aged



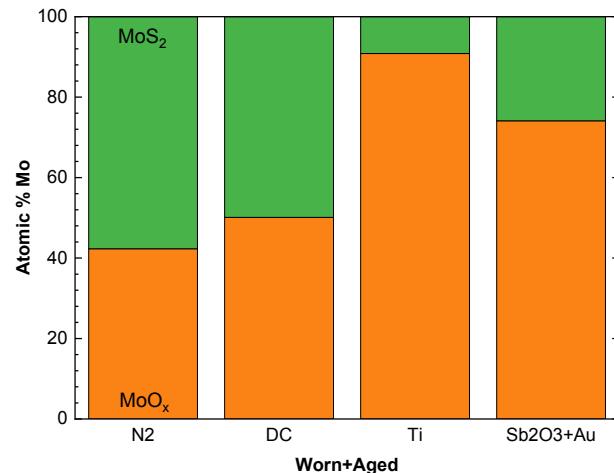
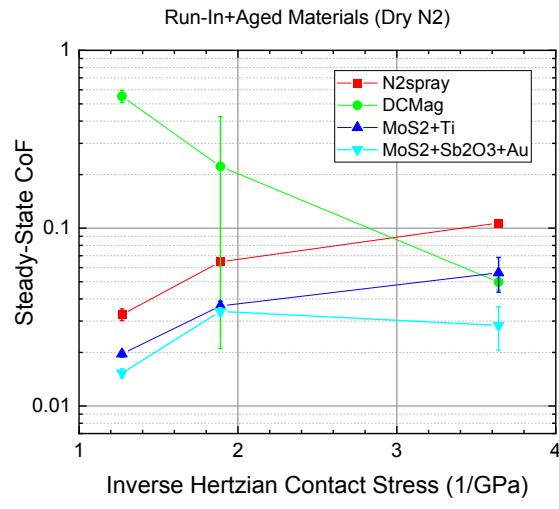
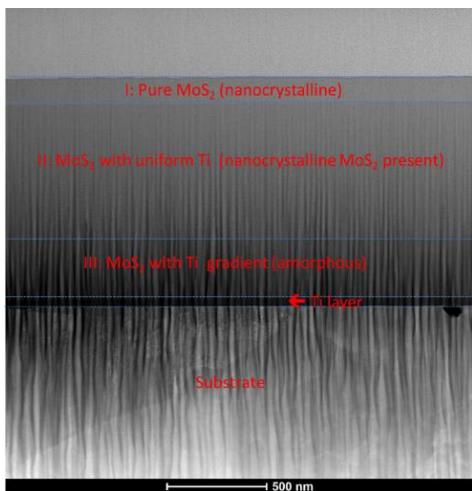


- MoS2-Sb2O3-Au generally lowest friction as deposited and behaved decently after aging
- MoS2-Ti films improved the most after aging
- Sprayed films & DC mag (both pure MoS2) relatively consistent but generally increasing after aging with DC mag exhibiting failures at higher loads

# MoS<sub>2</sub> Aging Series Takeaways



- Generally, doped films maintained lower steady state friction coefficients, and improved after aging/run-in prior to aging compared to undoped films
- Undoped films generally exhibit lowest initial friction behaviors compared to doped
- Undoped films also exhibit best oxidation resistance, likely due to lower reactivity of more crystalline materials at film surface (akin to run-in)
- Results suggest it is possible to sputtered MoS<sub>2</sub> films with structure that can resist oxidation & minimize initial friction via surface modification

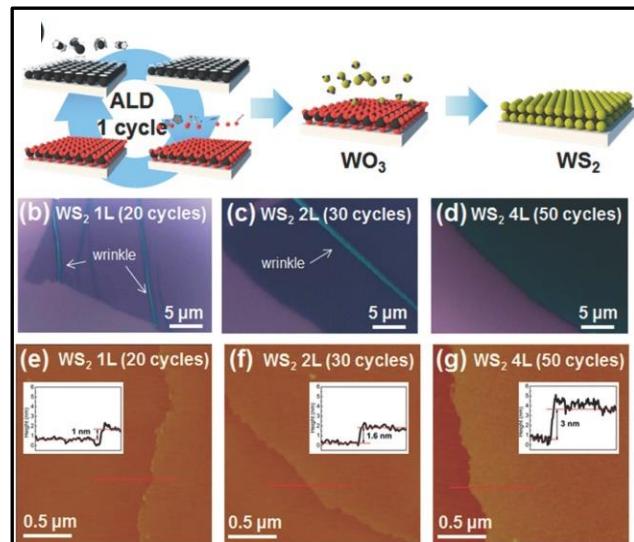
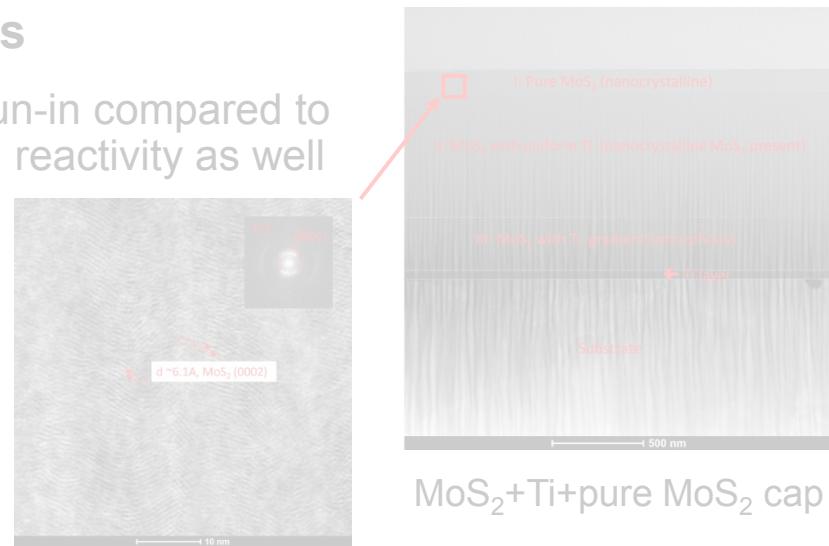
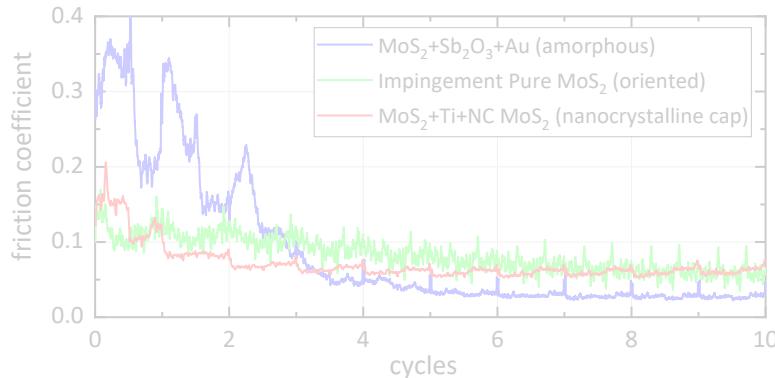


# Efforts to Reduce Run-In: Structure/Composition



## PVD nanocrystalline & composite films

- Films with NC structures exhibit reduced run-in compared to amorphous films (composites); likely reduced reactivity as well



## MoS<sub>2</sub> ALD at LBNL Molecular Foundry

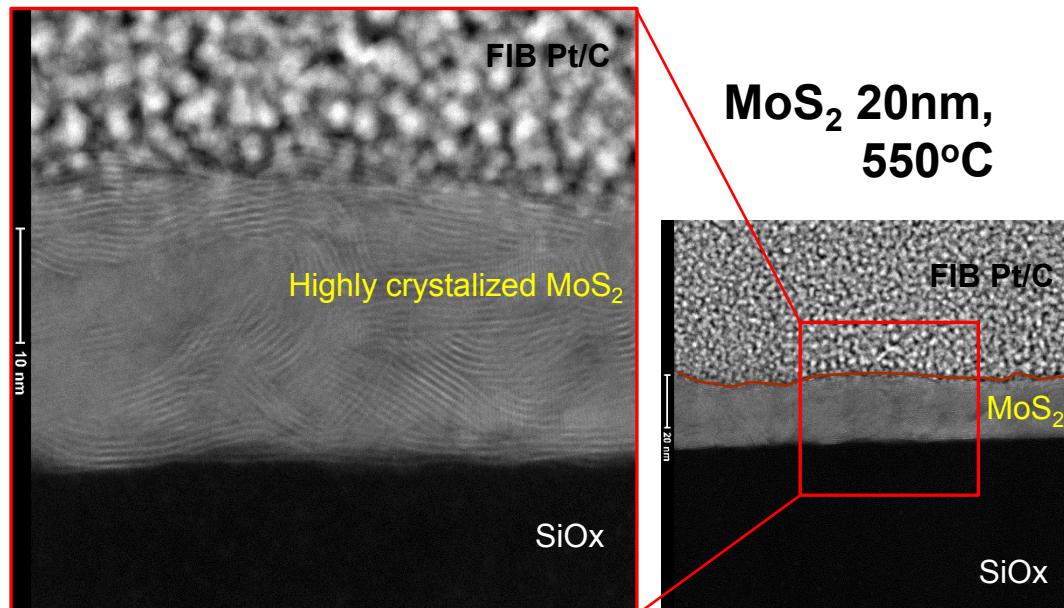
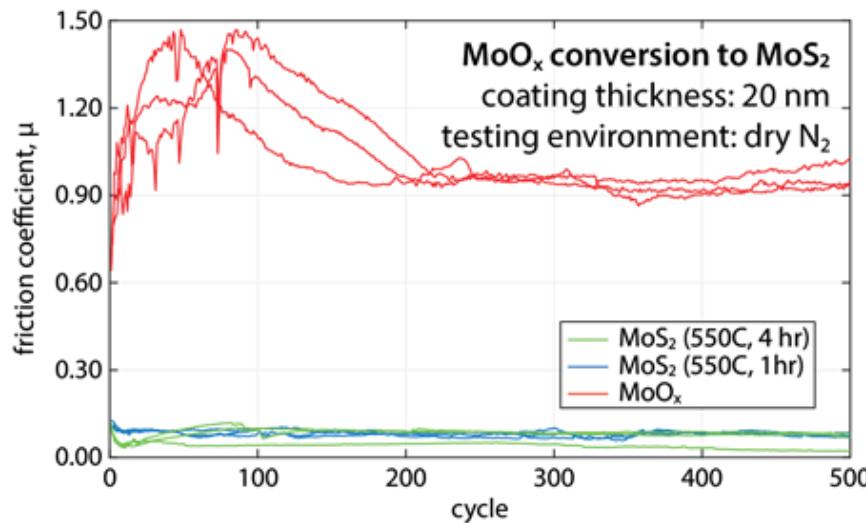
- Developing ALD MoS<sub>2</sub> freestanding and capped films for aging resistance
- MoO<sub>x</sub> conversion with Moly hexacarbonyl precursor and H<sub>2</sub>S conversion
- Challenges remain in optimizing synthesis to promote full conversion at low enough temperatures

# ALD MoS<sub>2</sub> Growth & Conversion



## Deposition/Conversion conditions

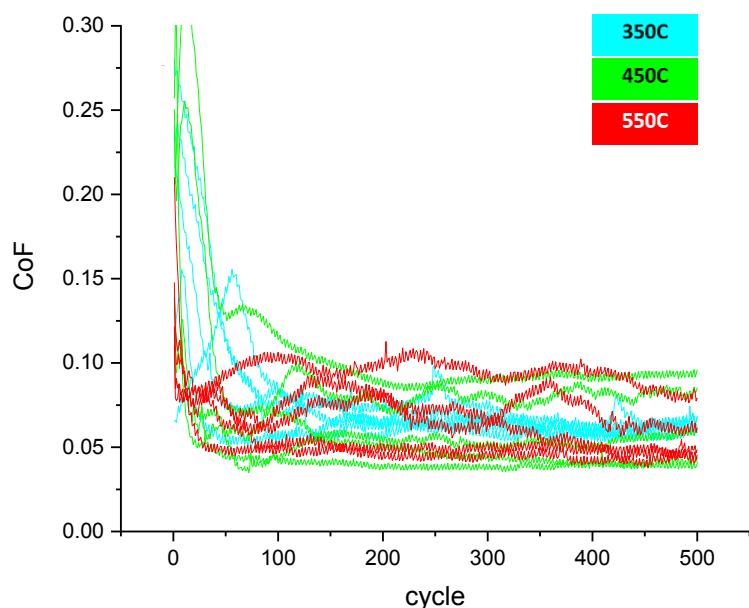
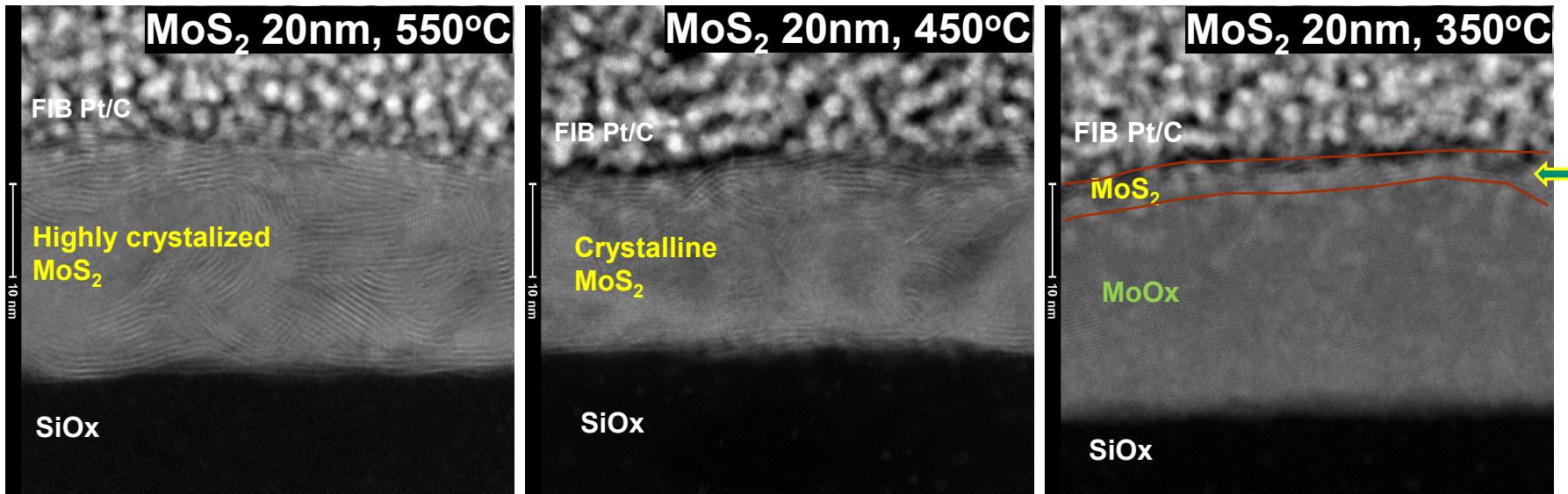
- Moly-hexacarbonyl precursor (AO plasma with water?), ???C for ???hr producing 20nm thick MoO<sub>x</sub> on SiO<sub>x</sub> wafer
- Initial conversion conditions: 550C for 1-4 hrs, flowing H<sub>2</sub>S/Ar mixture gas



## TEM/friction observations

- Pre/post friction data shows clear change in friction behavior from oxide to sulfide
- TEM confirm presence of fully converted MoS<sub>2</sub> at 550C conversion temperatures

# Conversion at lower temperatures?



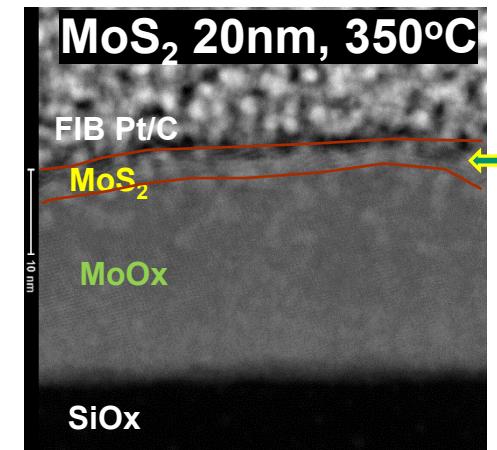
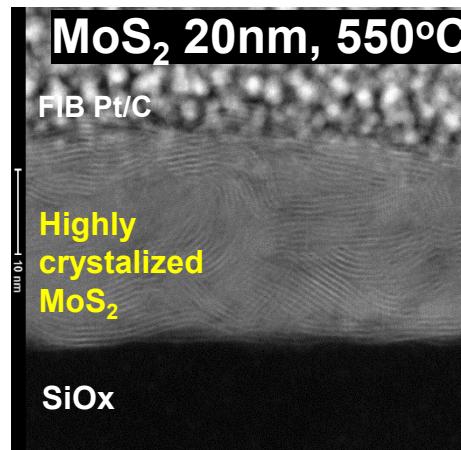
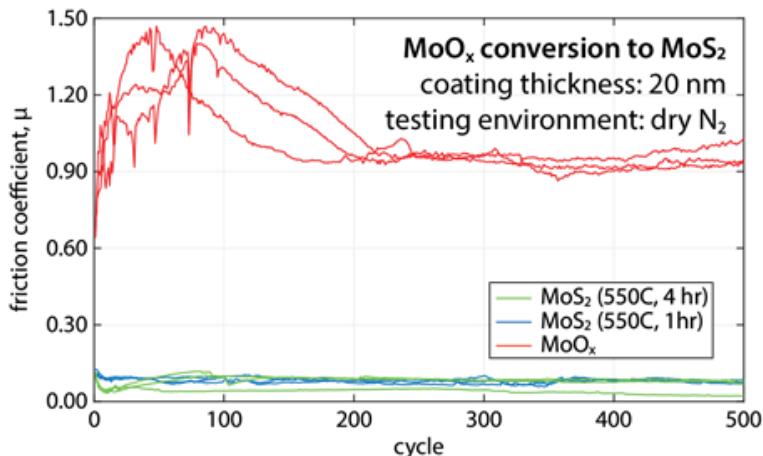
## Observations

- Friction behavior at all conversion temps indicates presence of MoS<sub>2</sub>
- TEM reveals complete conversion at  $\geq 450$  C
- For 350C conversion, only top few nm have converted to MoS<sub>2</sub> (still providing  $\sim 500$  cycles of low friction)

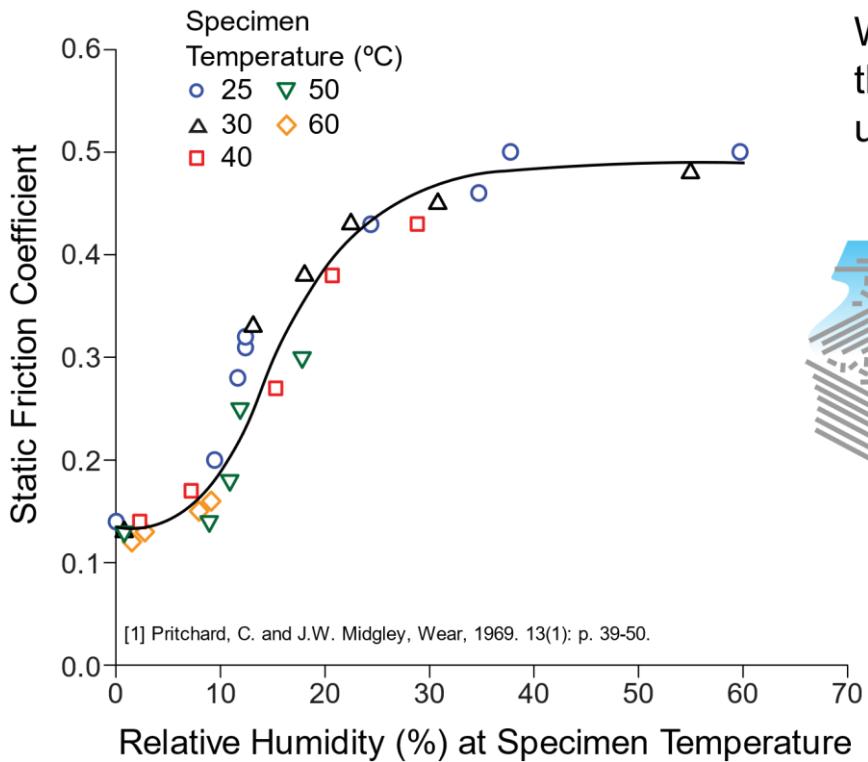
# MoS<sub>2</sub> ALD Takeaways



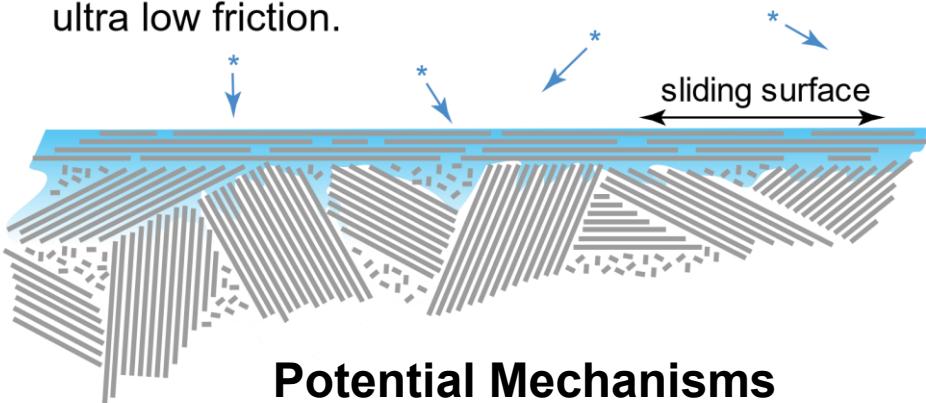
- MoO<sub>x</sub> growth and subsequent conversion is a relatively simple ALD process that can produce thin films with excellent friction behavior
- Complete conversion of 20nm oxide observed at 450C and above after 1 hr
- Incomplete conversion at 350C suggests kinetics not adequate, yet thin layer of MoS<sub>2</sub> still provides lubrication
- Additional work required to understand if conversion is possible at lower temperatures to enable deposition on work parts (i.e. steels)



# Fundamental Interactions with Water



Water adsorption and diffusion throughout a surface layer interrupts ultra low friction.



## Potential Mechanisms

- Adsorption (polar bonding, capillary forces, edge interactions, etc...)
- Oxidation (H<sub>2</sub>O vs O<sub>2</sub>, high temp, etc...)

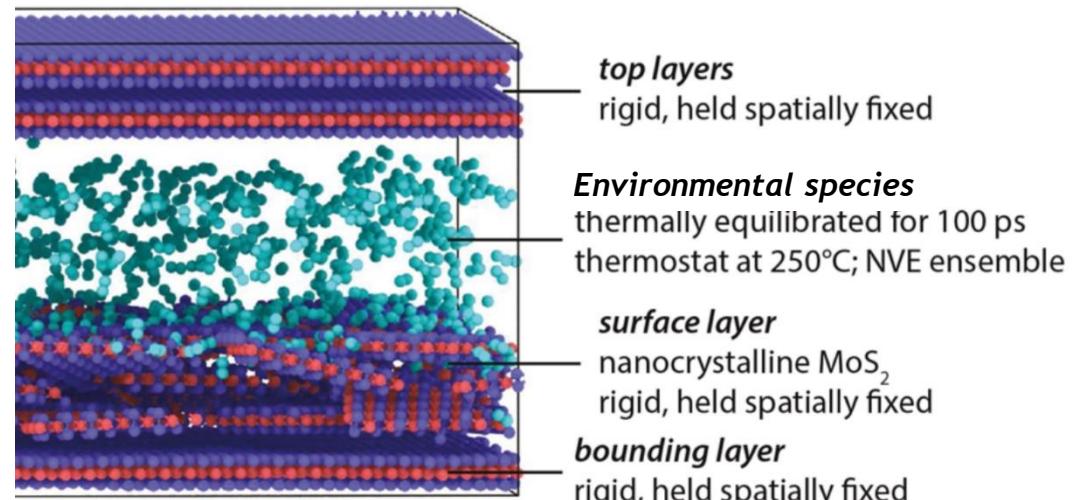
Little is known about how water fundamentally interacts and influences friction behavior in MoS<sub>2</sub>

# Structurally Driven Environmental Degradation of MoS<sub>2</sub>

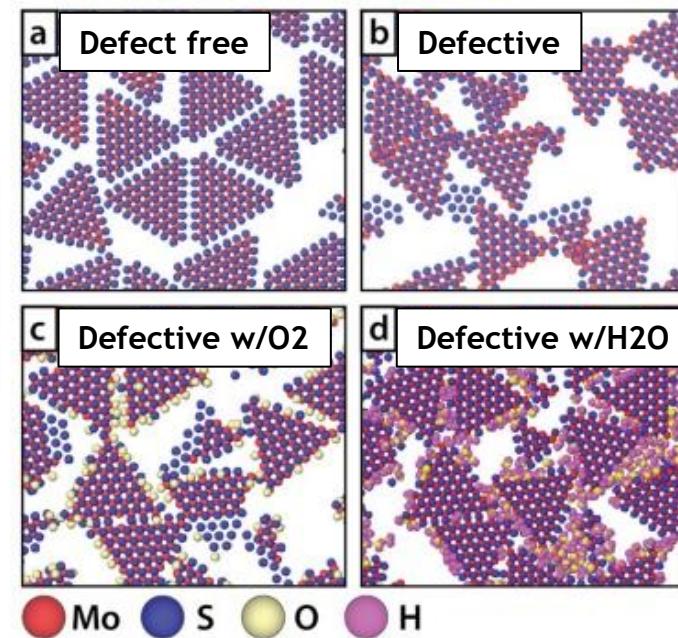
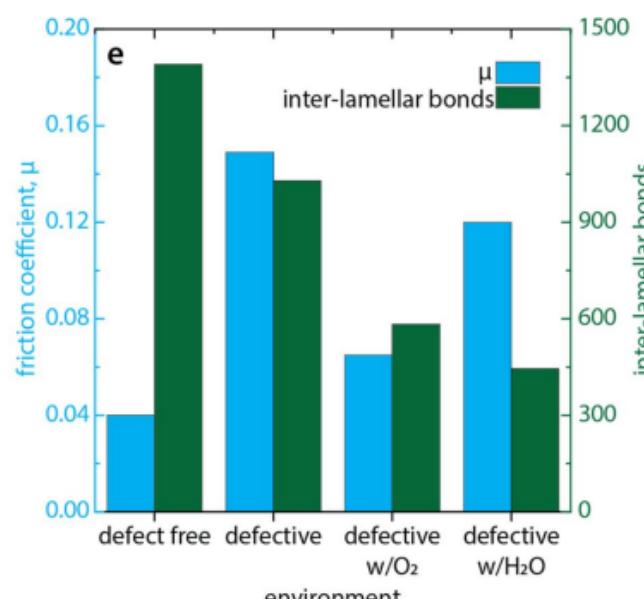


MD simulations show us a few things:

1. Sliding on pristine MoS<sub>2</sub> orders lamella and increases lamella size – low friction
2. O<sub>2</sub> and H<sub>2</sub>O passivate edge sites preventing coalescence lamella



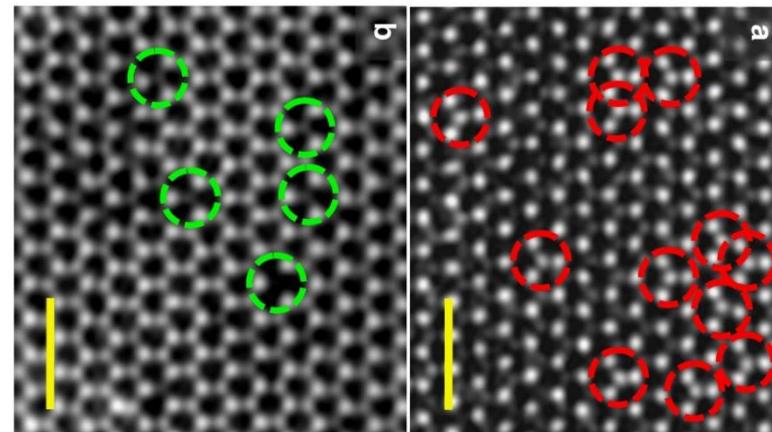
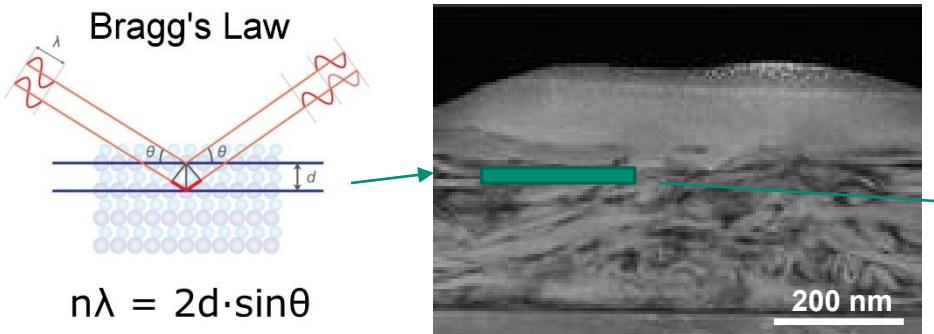
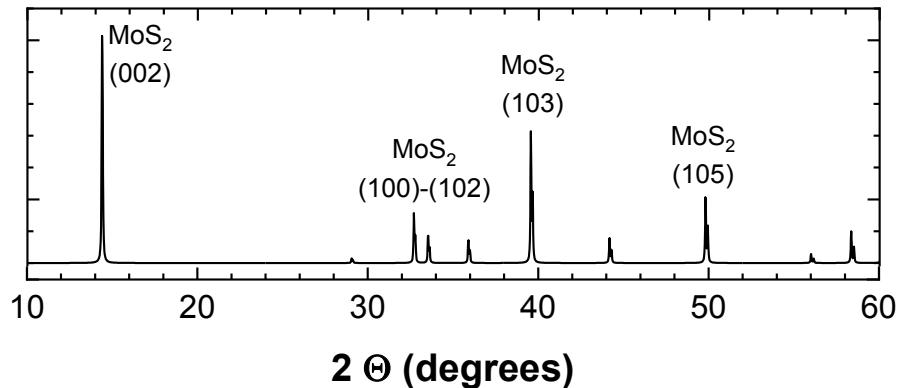
**Takeaway**  
**MD suggests**  
**changes to surface**  
**structure through**  
**environmental**  
**interactions dictate**  
**friction**



# Diagnosing Surface Structure is Difficult



## X-Ray Diffraction



STEM-ADF (Hong et al. Nat. Comm. 2015)

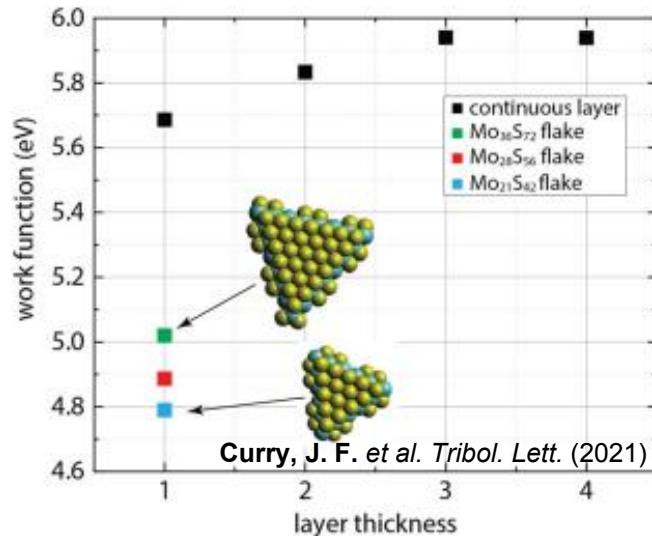
XRD can easily prove crystallite thickness, not width and depth

Existing methods for characterizing surface structure and defects (i.e. XRD, TEM, STM, etc) still make it difficult to assess near surface crystallite distribution and defect density.

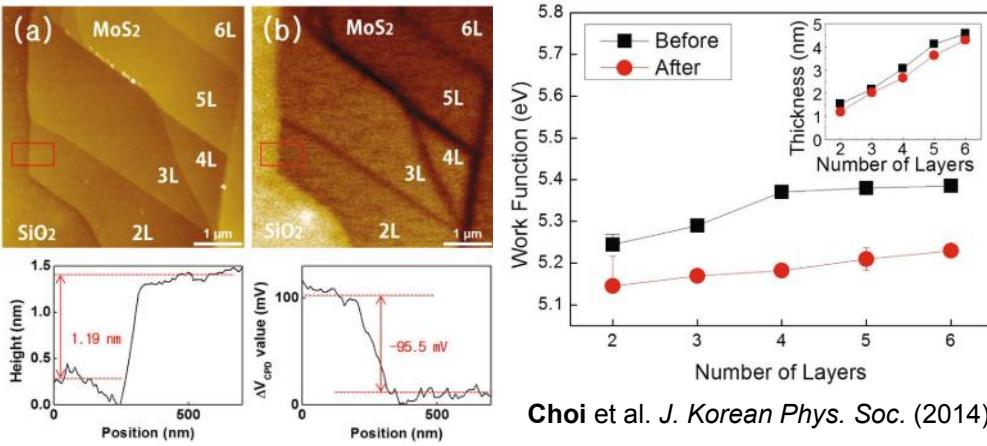
# Using work function to observe changes in microstructure



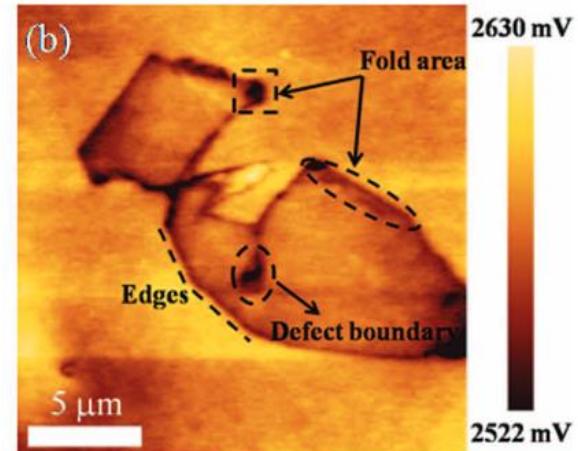
- Work function is a property of the surface
- Can be measured from KPFM, PEEM in combination with UPS
- Scales with number of layers
- Scales with the size of MoS<sub>2</sub> lamella



## Role of thickness & adsorbates

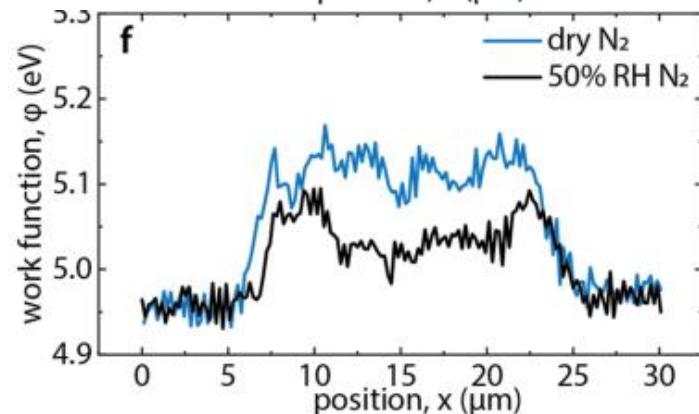
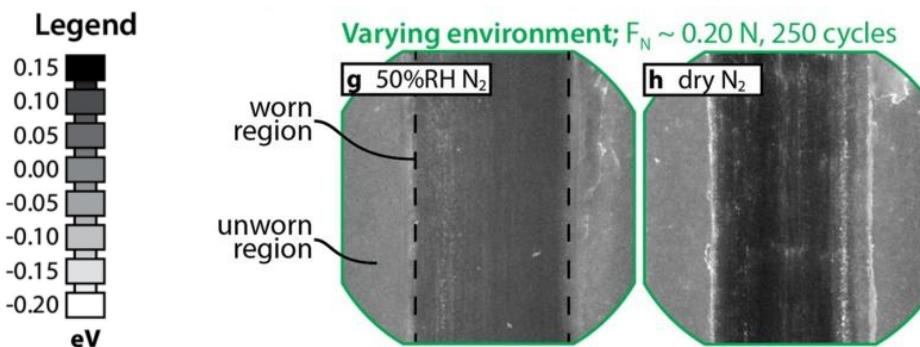
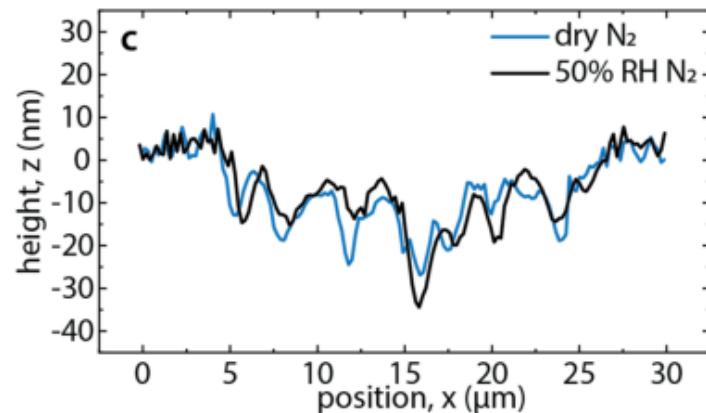
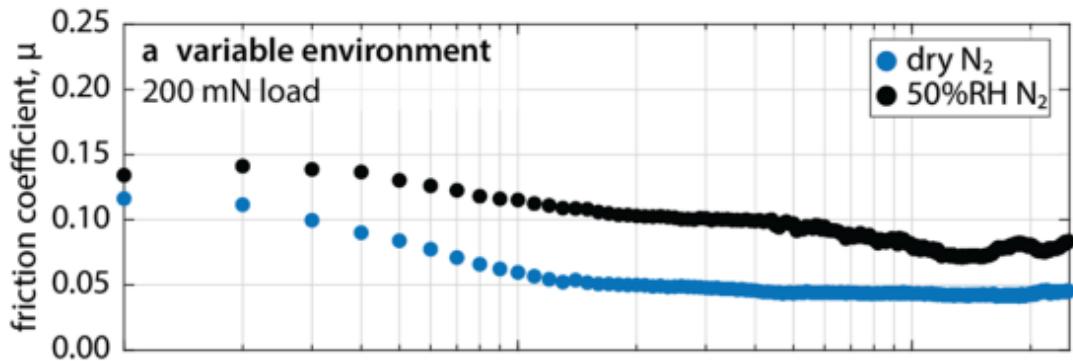


## Role of grain boundaries & edges



**Takeaway: Work-Function can be used to assess changes in surface microstructure**

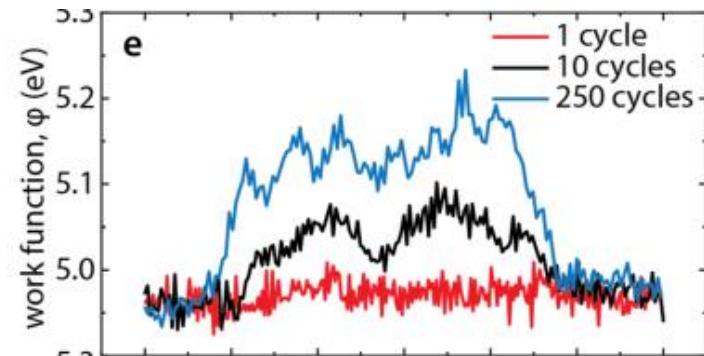
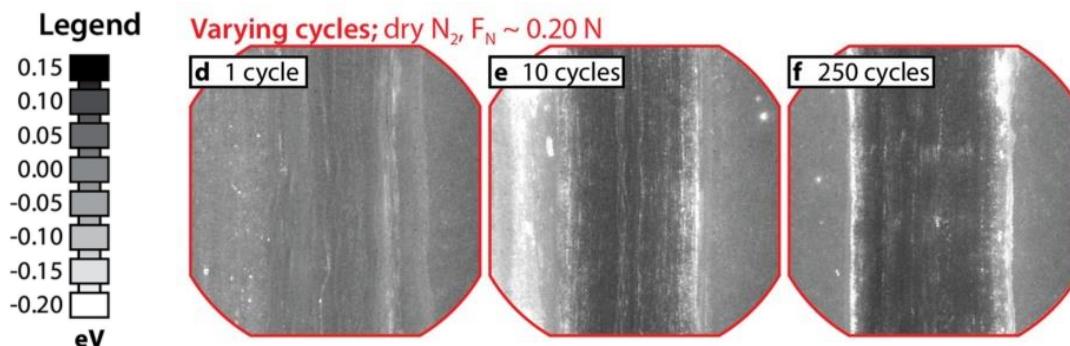
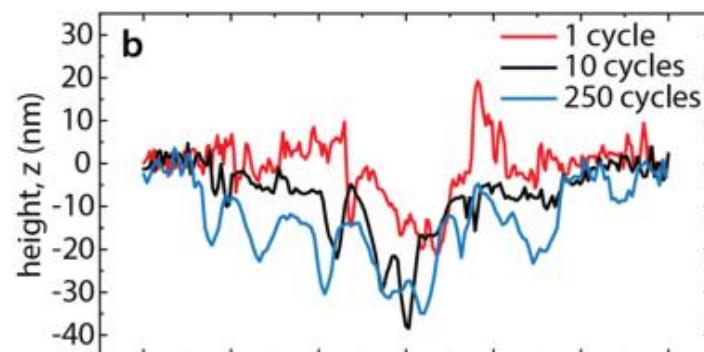
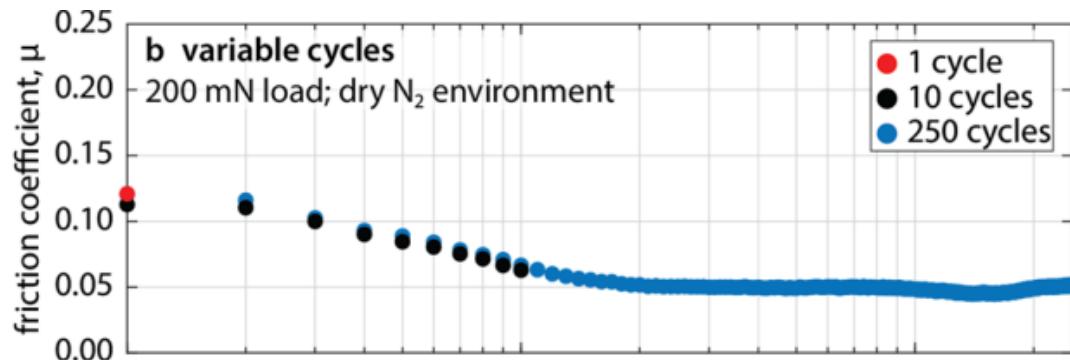
# Friction response driven by changes in structure



KPFM and PEEM indicate that shear in presence of  $\text{H}_2\text{O}$  still increases work function compared to bulk, but less than sliding without  $\text{H}_2\text{O}$

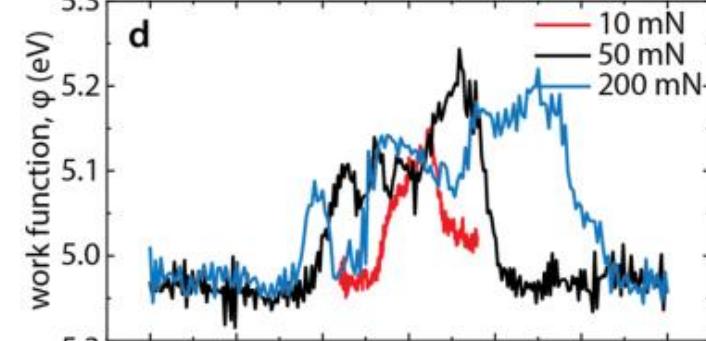
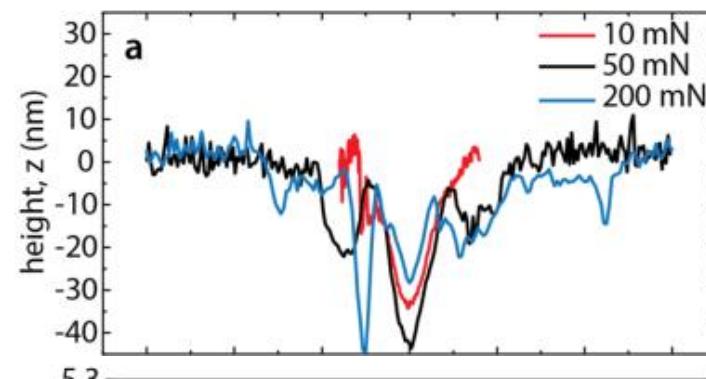
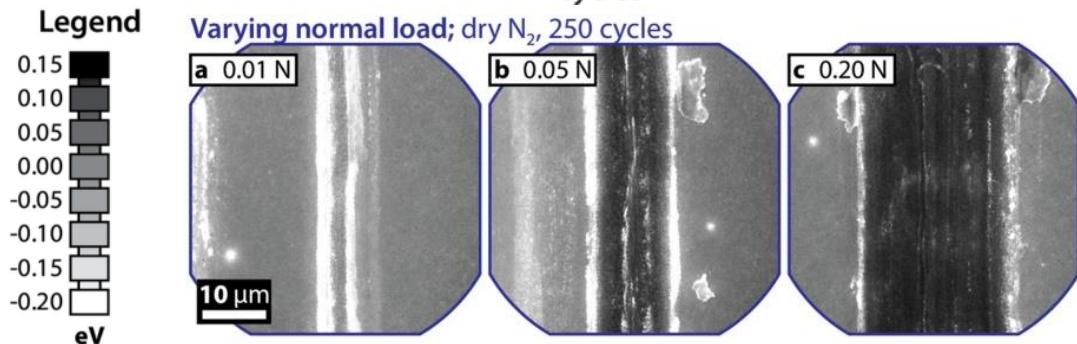
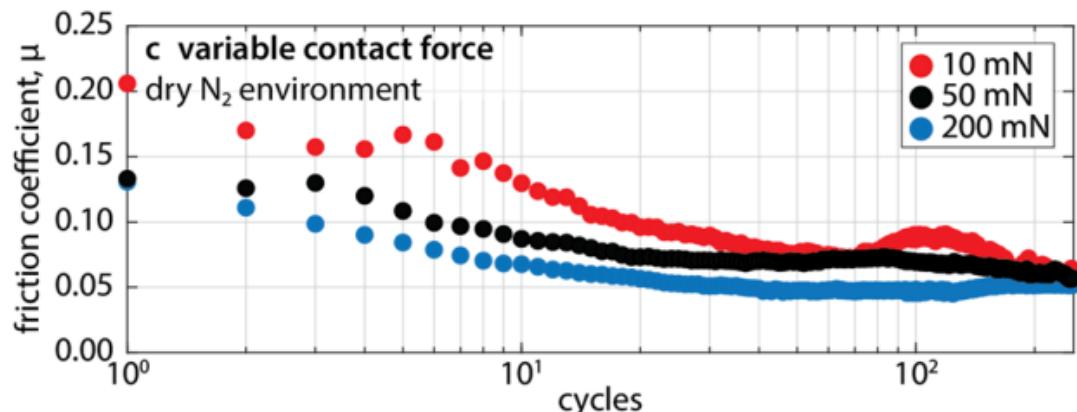
**Takeaway:  $\text{H}_2\text{O}$  decreases work function by inhibiting formation of large defect free lamella**

# Friction response driven by changes in structure



**Takeaway #2: The transition to low friction “run-in” is a result of shear combining and reorienting lamella**

# Friction response driven by changes in structure

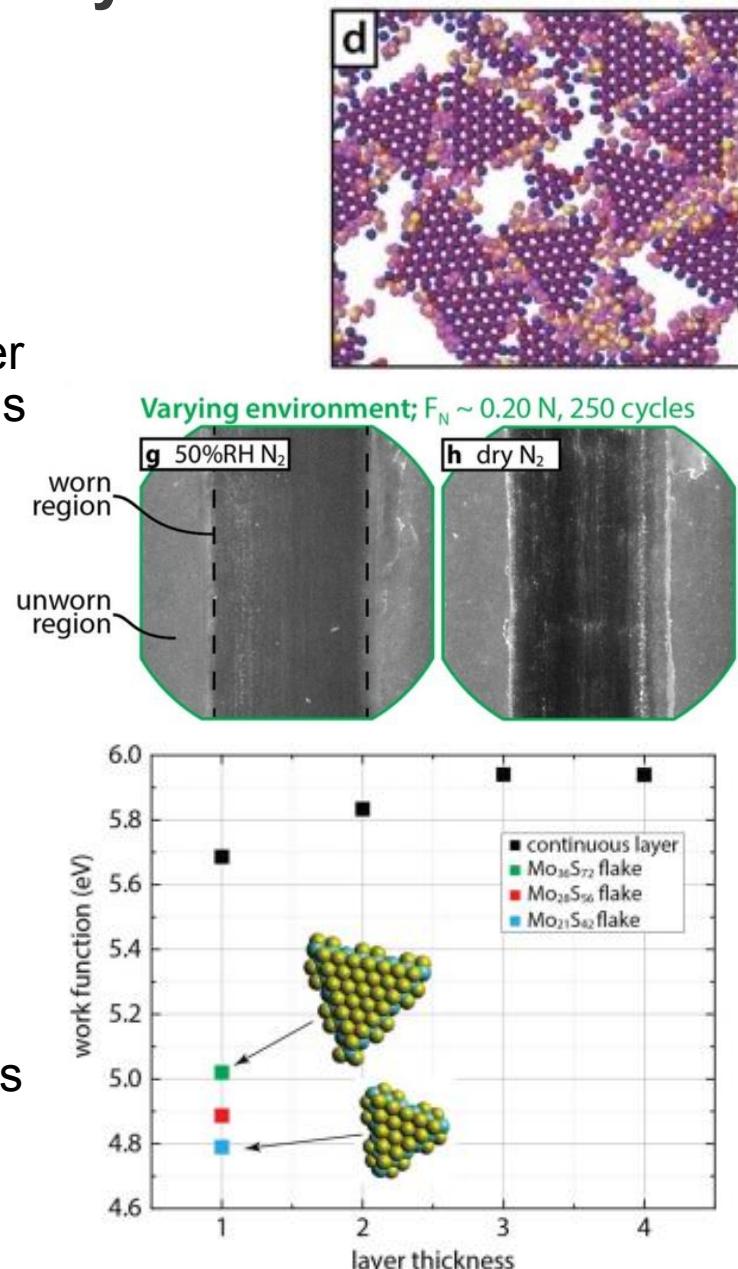


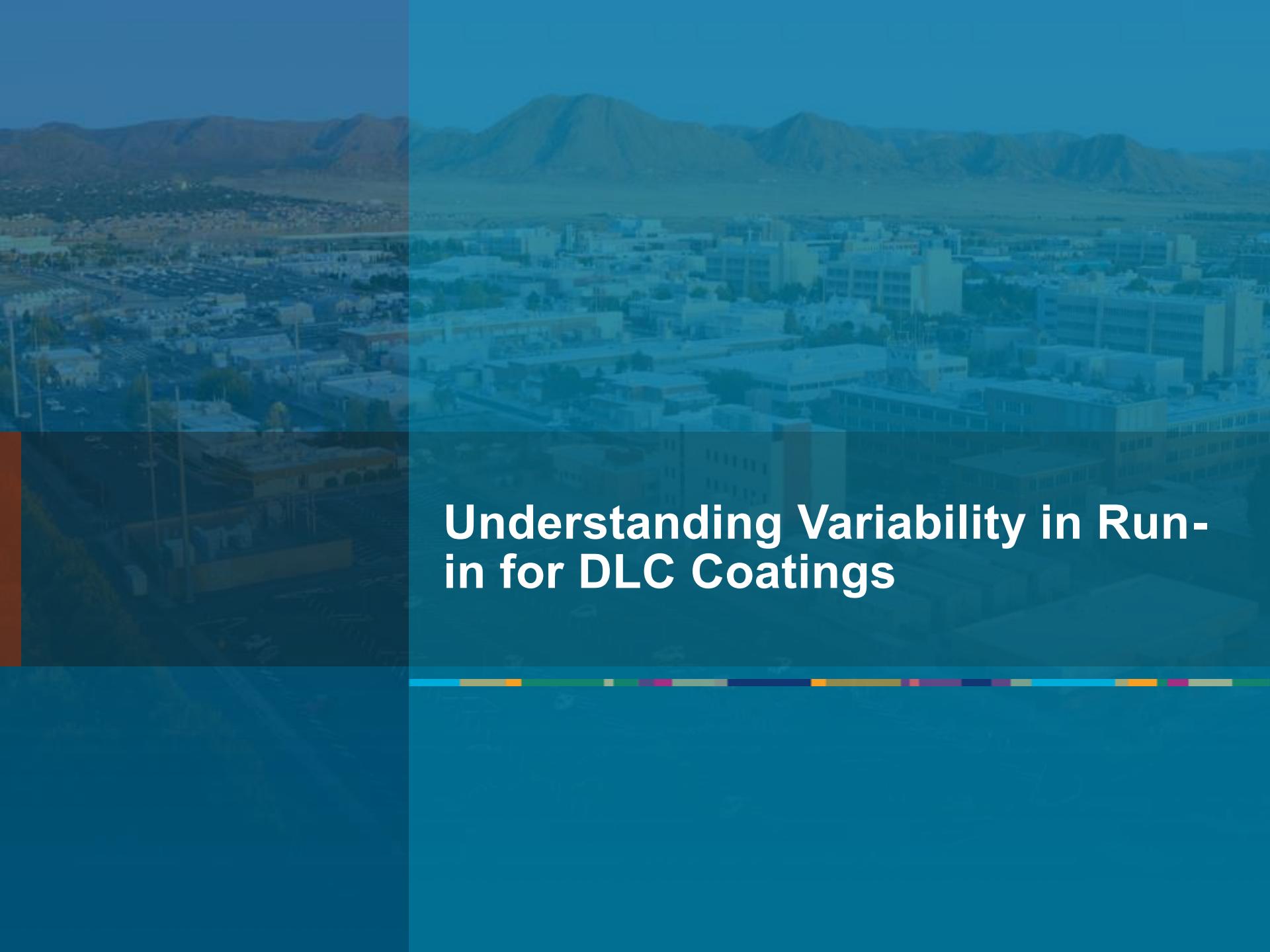
*Takeaway #1: Increasing contact pressure forms larger lamella*

# Work Function on MoS<sub>2</sub> Take-Aways



- MD simulations suggest water interaction/agglomeration with edge sites prevents formation of long range order MoS<sub>2</sub>
- Structural degradation (smaller flake size, higher defect density) due to environmental interactions leads to increased friction
- Work function can be used to probe changes in structure at the surface at macroscale
- DFT and literature show that higher work functions are related to larger crystallites and higher layer counts (thickness) of MoS<sub>2</sub>
- Lower work functions were observed in wear scars associated with higher friction conditions (humidity, low load/cycle count) likely due to less ordered structures

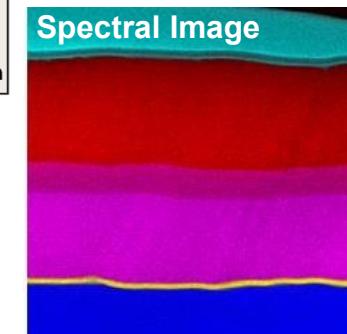
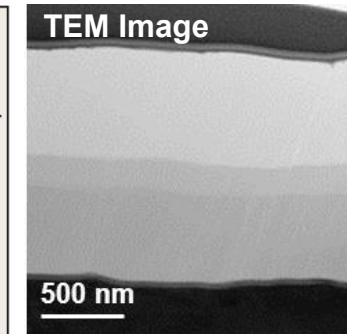
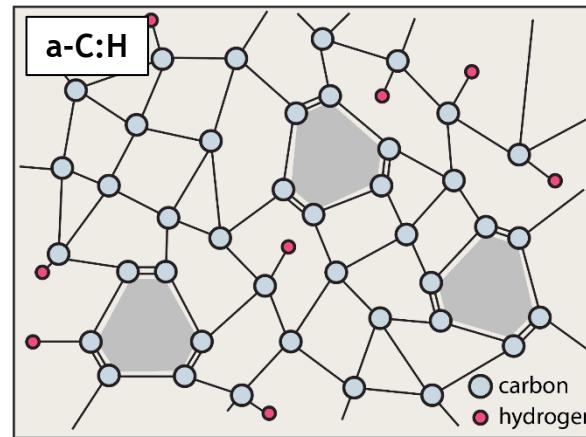
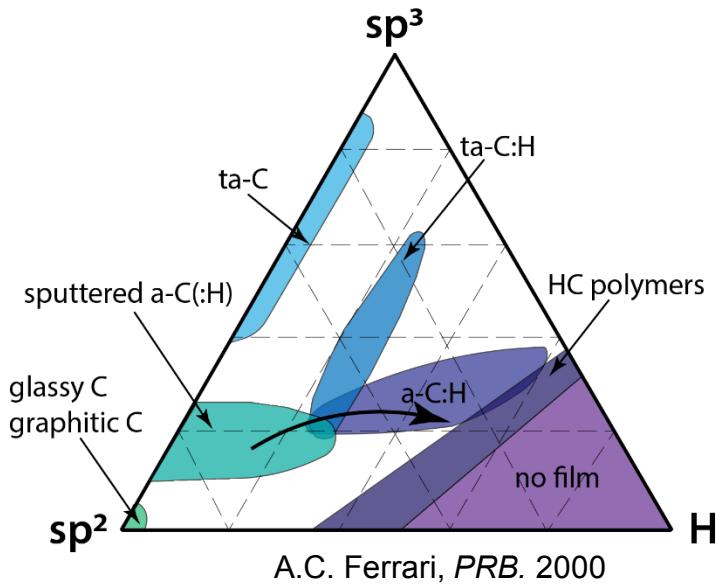




# Understanding Variability in Run-in for DLC Coatings



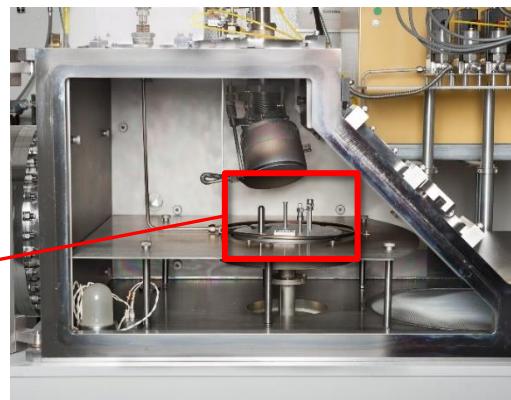
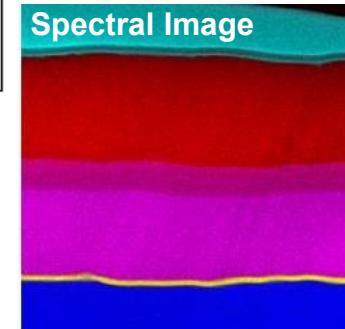
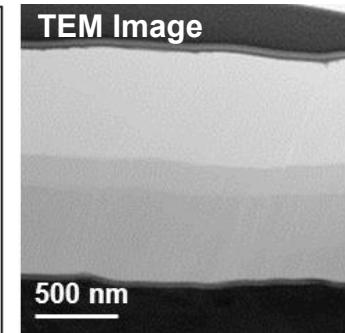
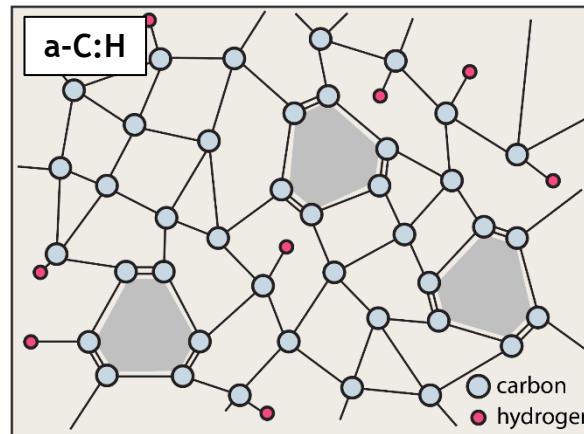
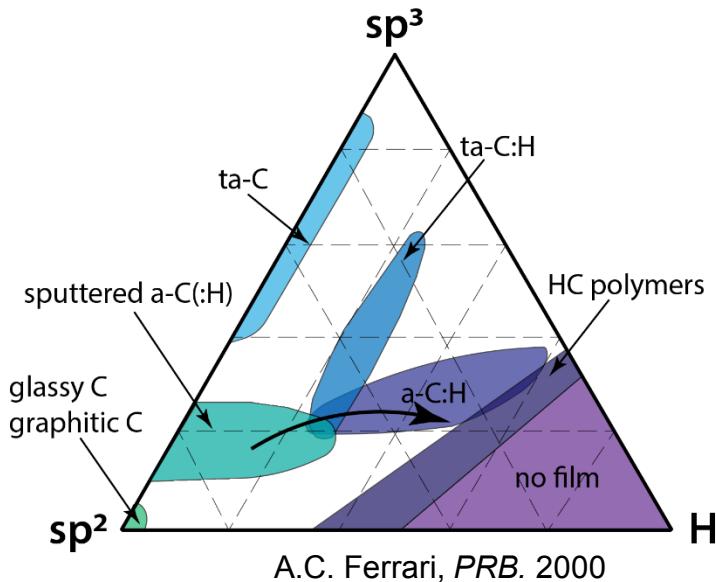
# Diamond-Like Carbon (DLC) Coatings



FIB Pt  
C  
Si+C  
Ti steel

- Amorphous network of sp<sup>2</sup>/sp<sup>3</sup>/H or other dopant

# Diamond-Like Carbon (DLC) Coatings



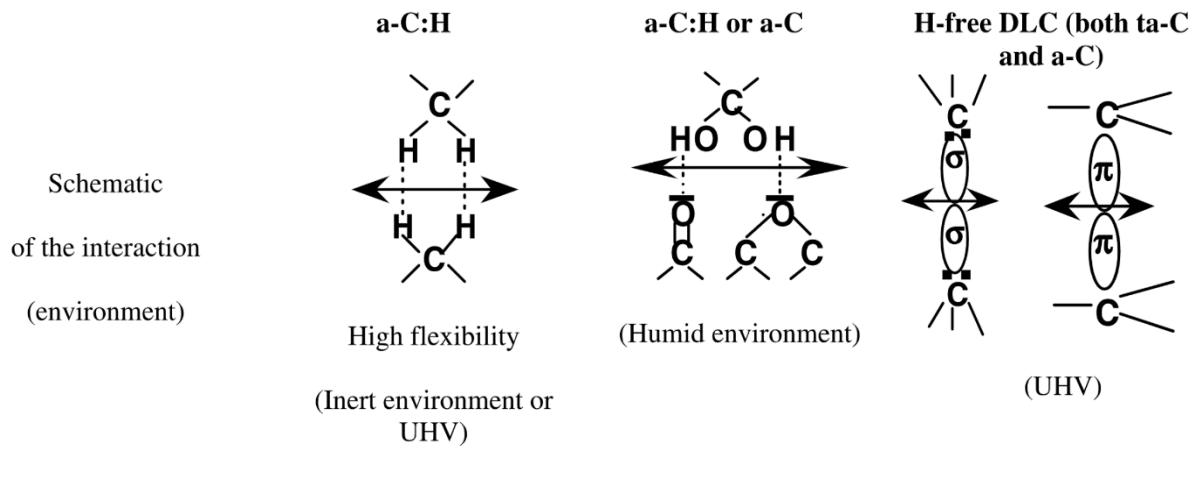
Images courtesy of Mike York, NSC

- Amorphous network of sp<sub>2</sub>/sp<sub>3</sub>/H or other dopant
- PECVD process decomposes precursor hydrocarbon gases to deposit carbon films
- Result in complex film compositions with varying tribological behavior...

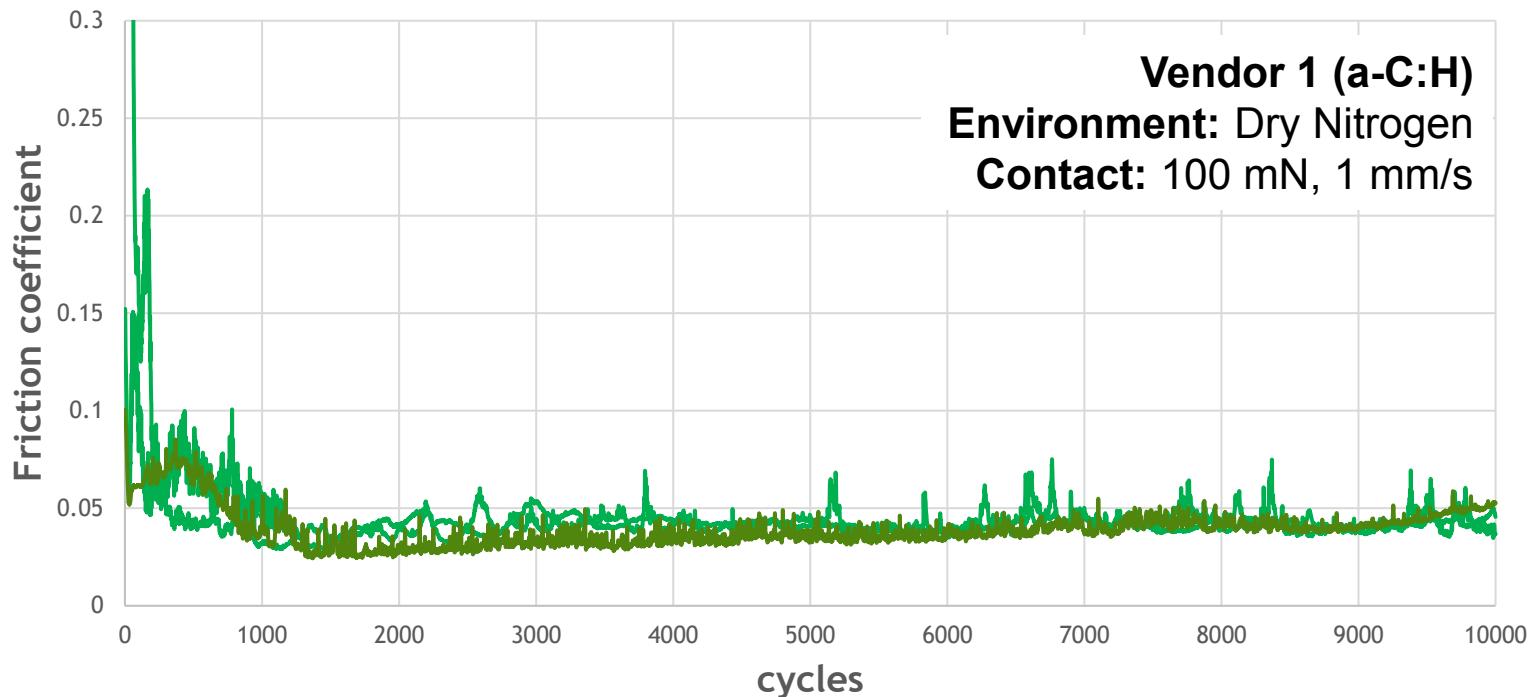
# Friction Behaviors in DLCs



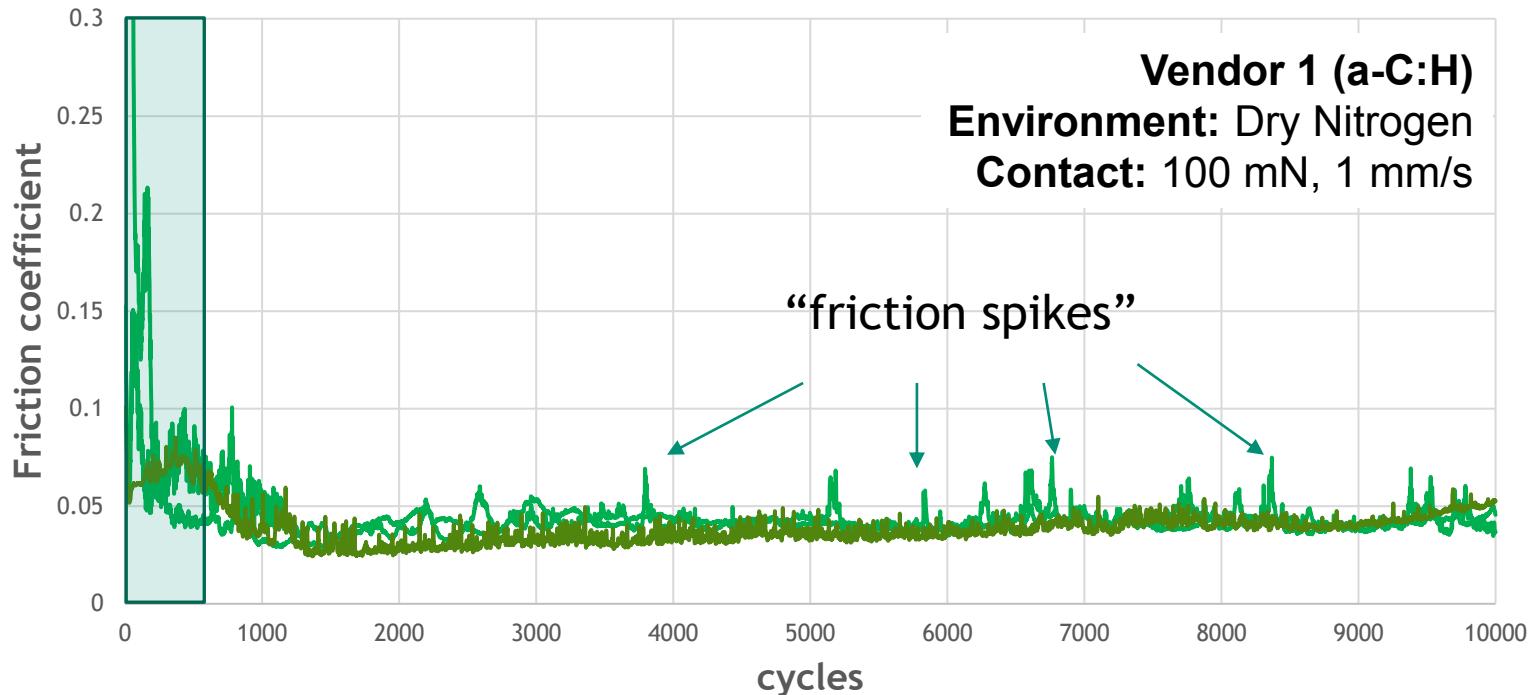
| solid lubricant                 | deposition methods                    | $\mu_{ss}$                | $F_n$       | environment   |
|---------------------------------|---------------------------------------|---------------------------|-------------|---------------|
| graphite ( $sp^2$ bonding)      | evaporation, pyrolysis of HC polymers | 0.2 - 0.5                 | 0.5 N - 1 N | dry $N_2/UHV$ |
|                                 |                                       | 0.1 - 0.2                 | 0.5 N - 1 N | humid air     |
| DLC (mixed $sp^2/sp^3$ bonding) | rf and dc sputtering, ion beam, CVD   | 0.6 - 0.7 a-C             | 10 N        | dry $N_2/UHV$ |
|                                 |                                       | <b>0.001 - 0.05 a-C:H</b> | 10 N        | dry $N_2/UHV$ |
|                                 |                                       | 0.1 - 0.2 a-C             | 10 N        | humid air     |
|                                 |                                       | 0.2 - 0.3 a-C:H           | 10 N        | humid air     |



# Typical DLC Friction Trace

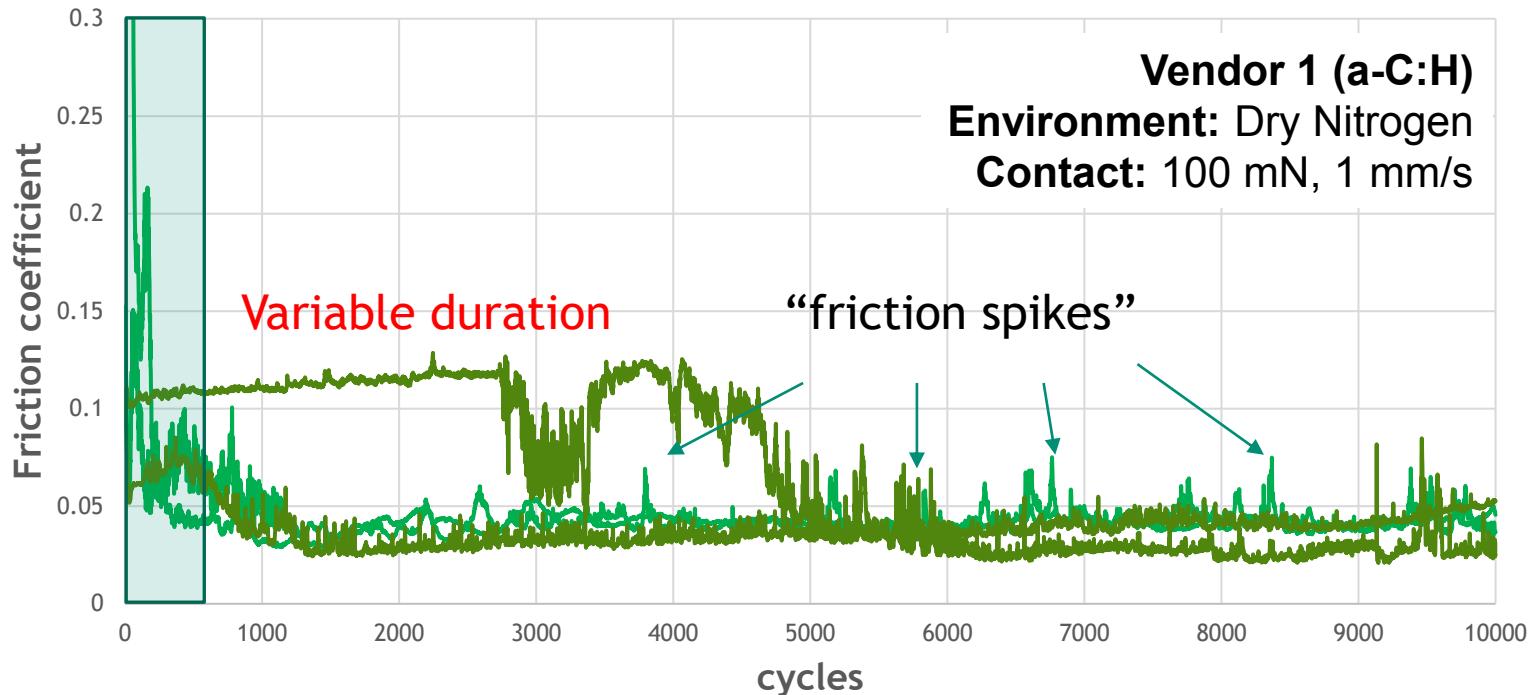


# Typical DLC Friction Trace



- Coatings exhibit low steady state friction and occasionally “spikey” steady state behavior
- Magnitude of initial friction may vary between  $\mu=0.10-0.30\dots$

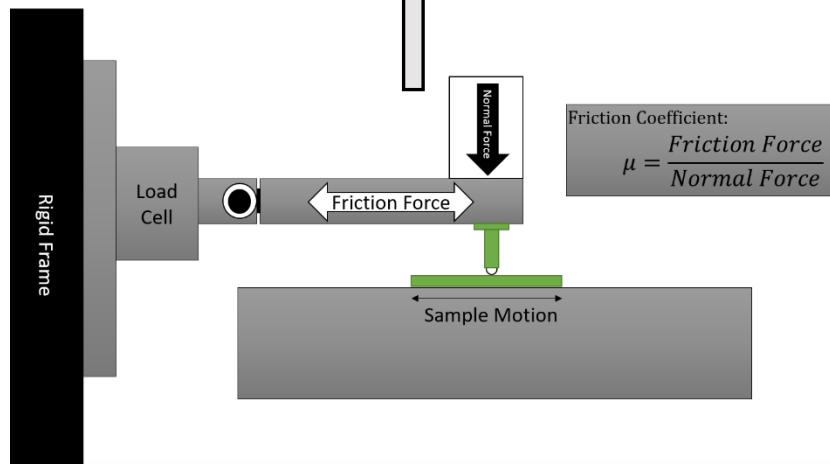
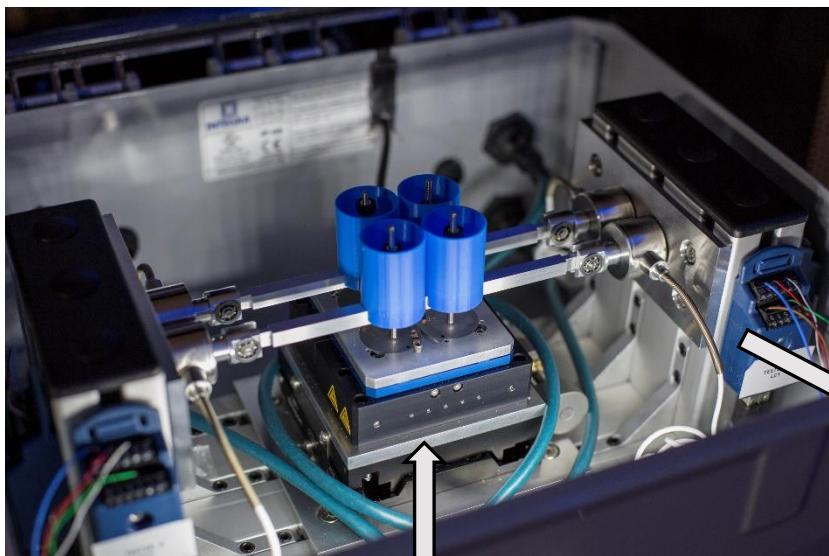
# Typical DLC Friction Trace



- Coatings exhibit low steady state friction and occasionally “spikey” steady state behavior
- Magnitude of initial friction may vary between  $\mu=0.10-0.30\dots$  run-in duration varies extensively as well

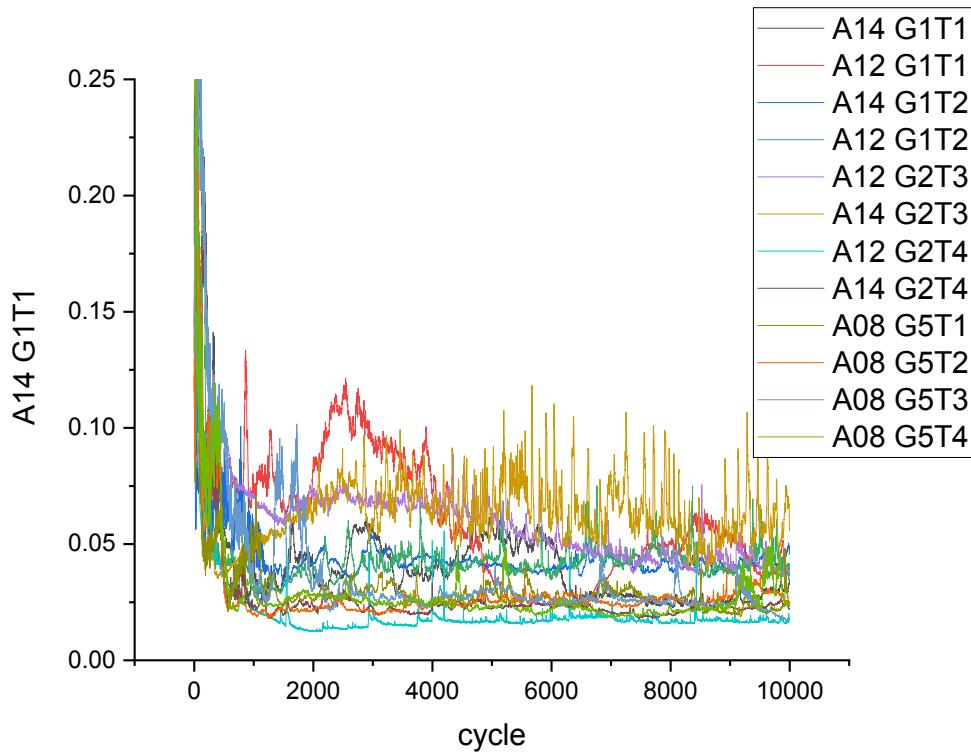
***Variability in run-in (like MoS<sub>2</sub>) can be a detriment to reliable operation***

# Brute Force – High Throughput Friction Testing



**16X simultaneous testing capacity, multiple environments**

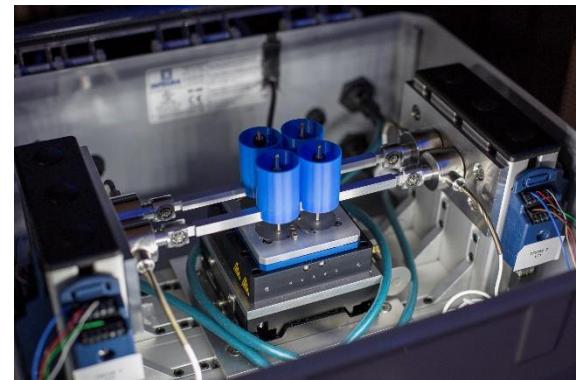
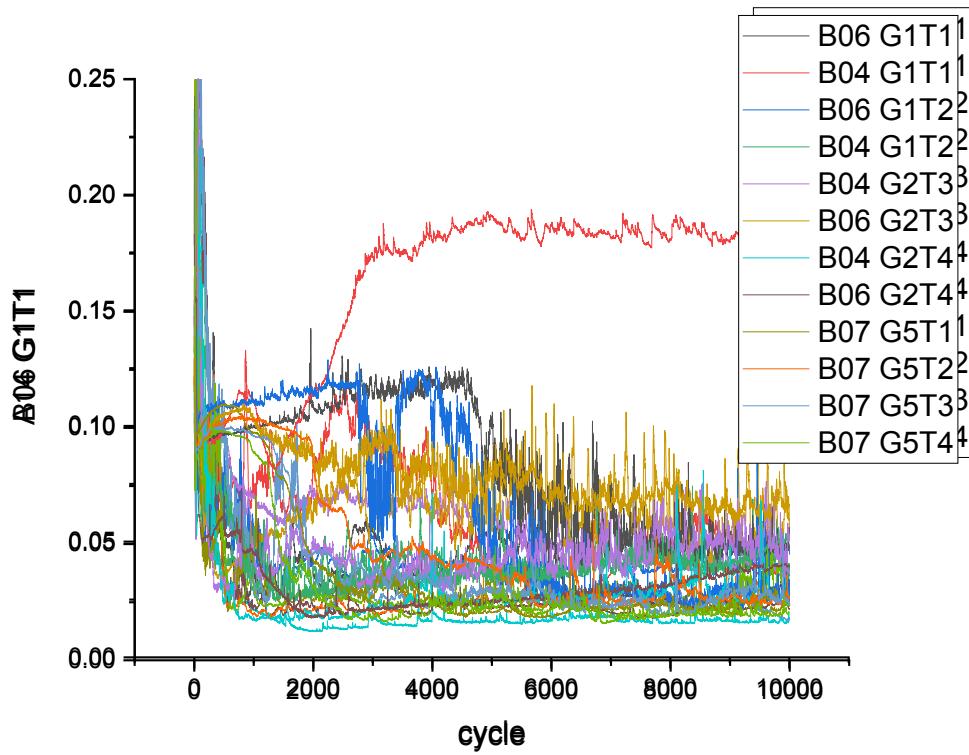
# HTT Case Study - Composition



## Test Specifics

- 5 DLC coatings; 4 samples each; 3-4 tests per sample @ 10,000 cycles
- 500 MPa (130 mN) max hertz contact pressure; 2mm stroke, 1 mm/s

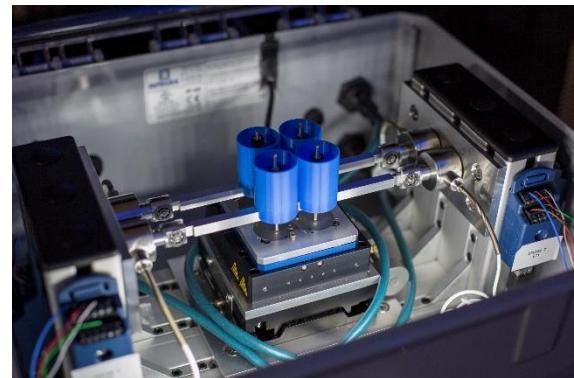
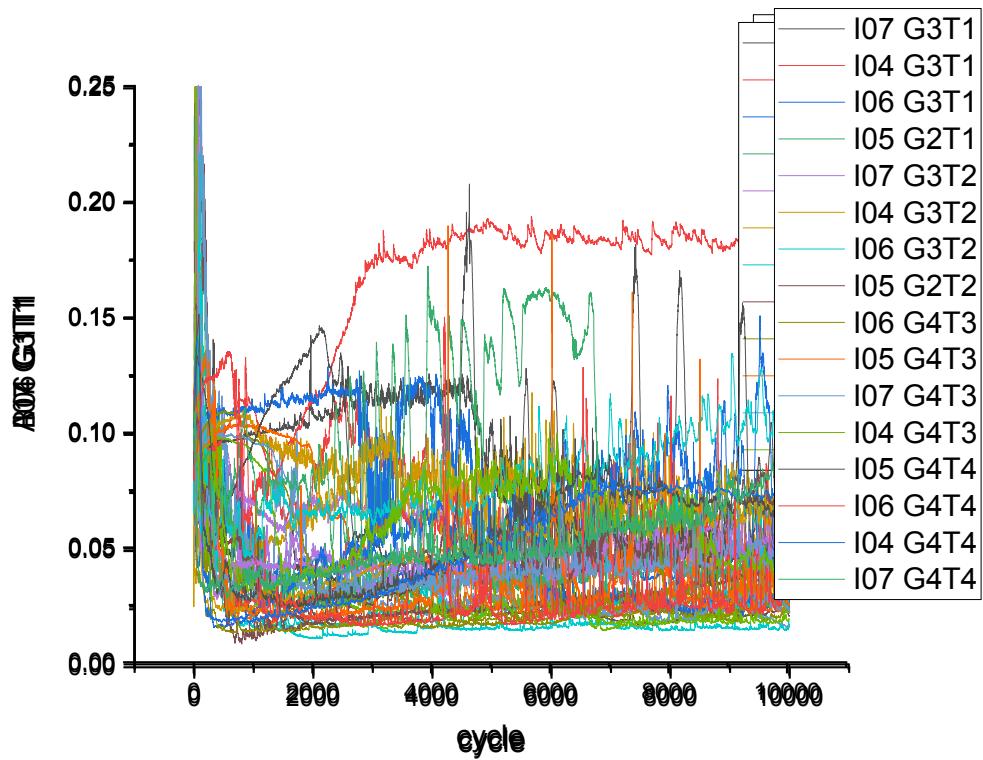
# HTT Case Study - Composition



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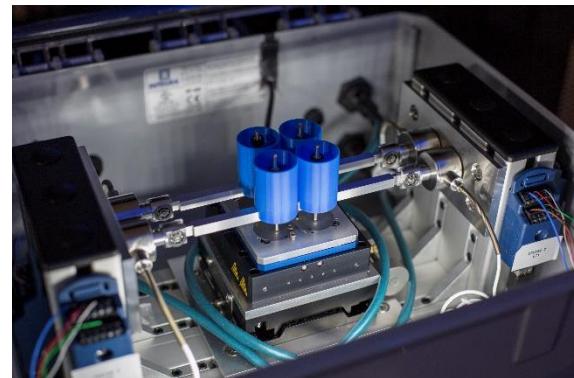
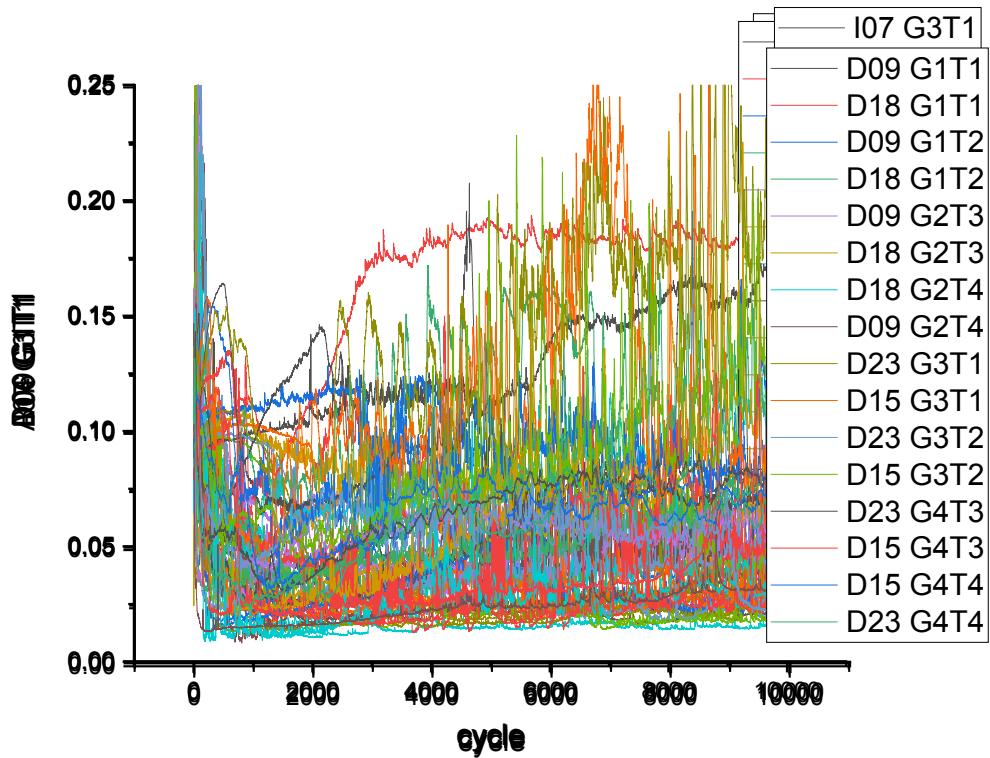
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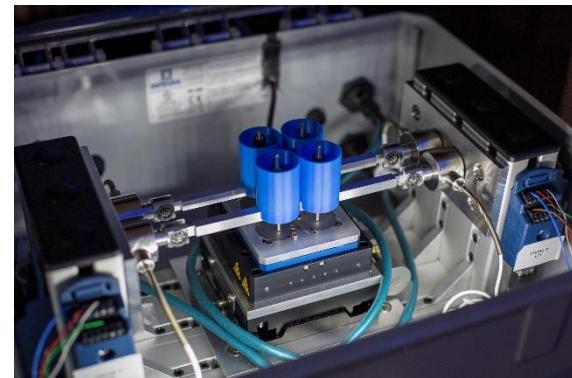
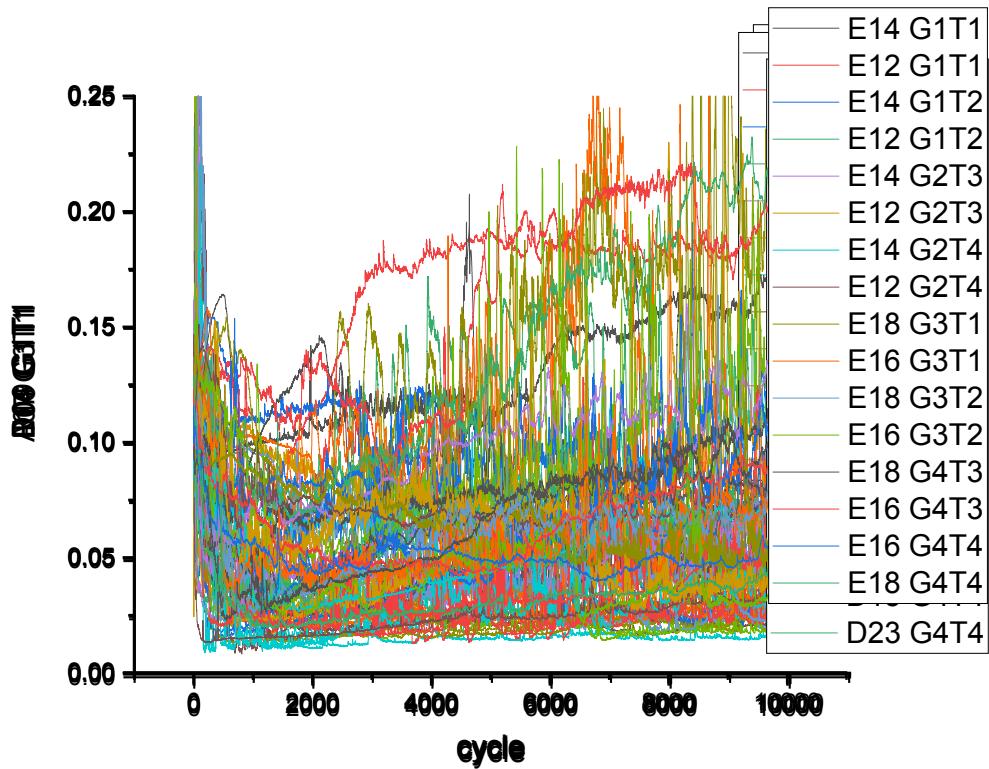
# HTT Case Study - Composition



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# HTT Case Study - Composition



## Test Specifics

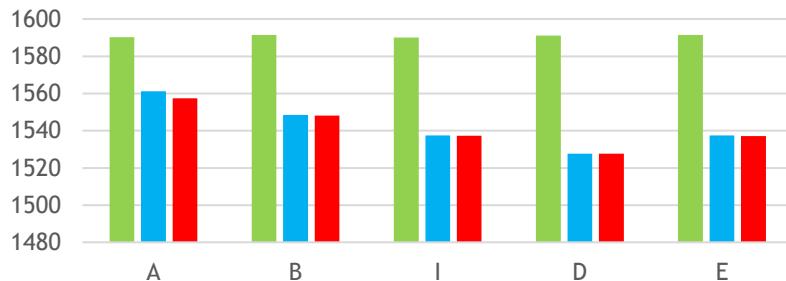
- 5 DLC coatings; 4 samples each; 3-4 tests per sample @ 10,000 cycles
- 500 MPa (130 mN) max hertz contact pressure; 1mm stroke, 1 mm/s

***~75 experiments running 10K cycles each, taking under 1 week to finish***

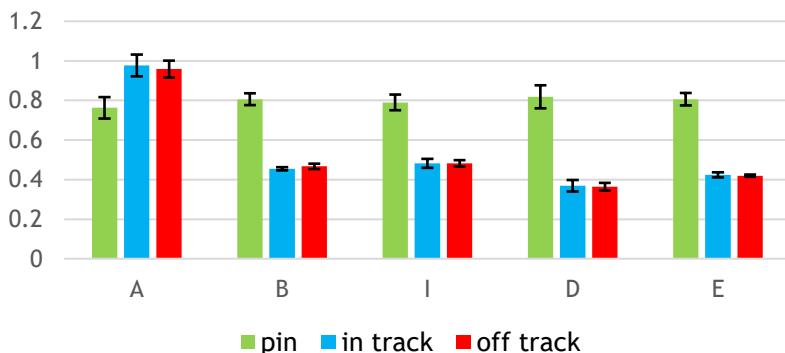
# HTT Case Study - Results



Raman G Peak Position



Raman ID/IG Intensity Ratios

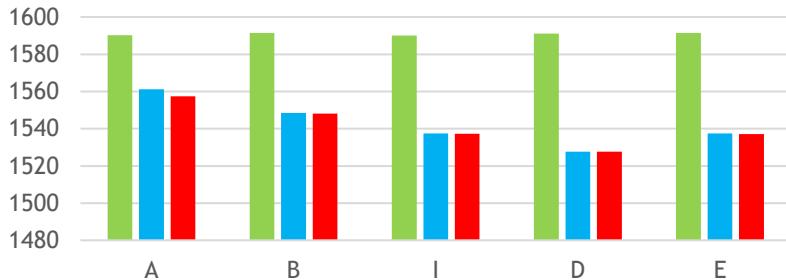


- In/out track Raman similar, exhibit differences unlike pin surface (unique from original film)

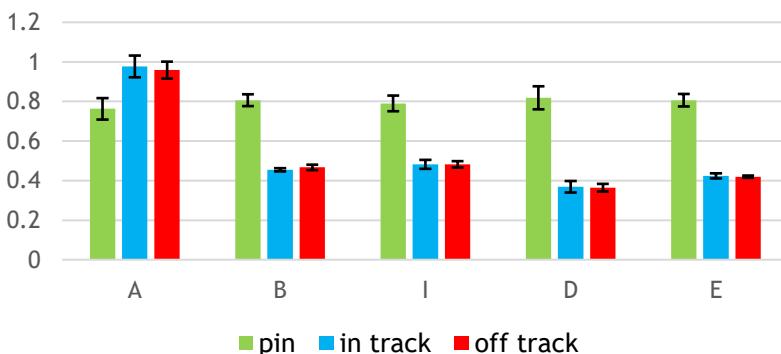
# HTT Case Study - Results



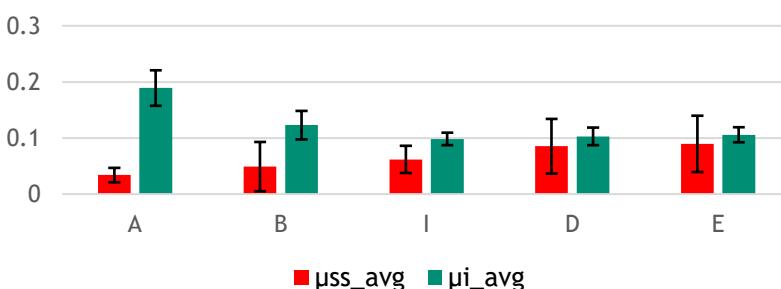
Raman G Peak Position



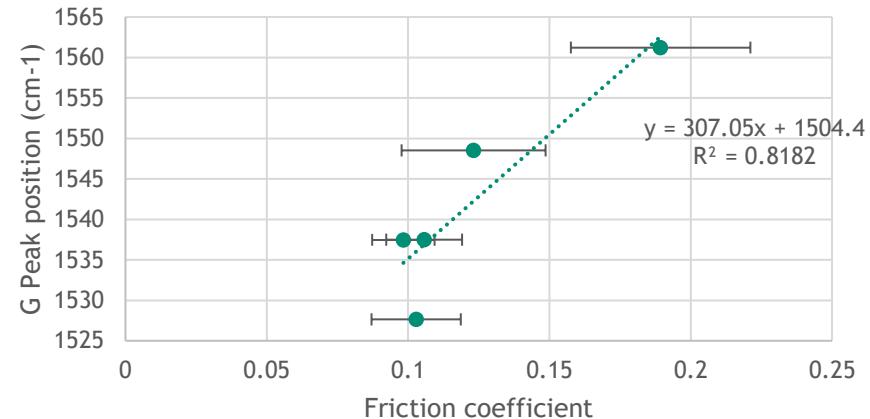
Raman ID/IG Intensity Ratios



Friction behavior



Initial Friction vs track G Peak Position



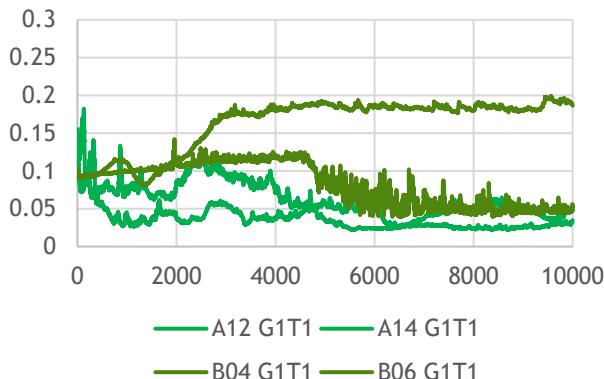
- In/out track Raman similar, exhibit differences unlike pin surface (unique from original film)

- Steady state and initial friction are inverse; correlate well with G peak pos & ID/IG of track

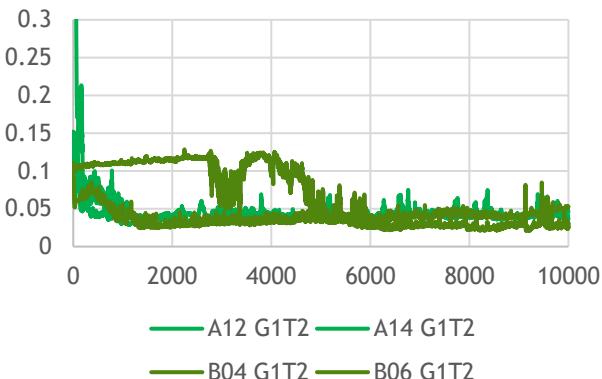
# Effects of Purge Time on Variability



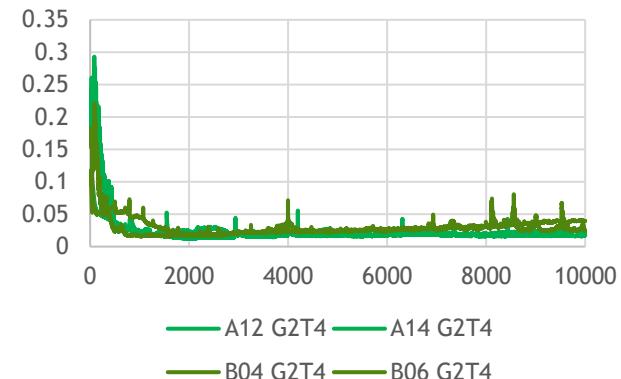
Group 1, Test 1, Module 1 - Purge Time 1hr



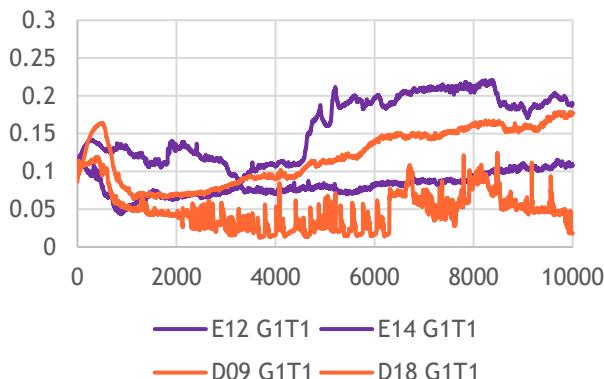
Group 1, Test 2, Module 1 - Purge Time 8hr



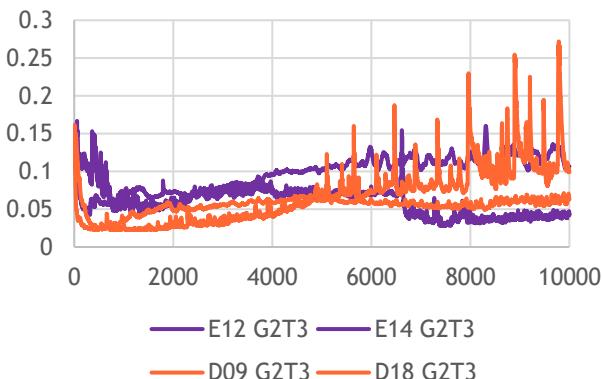
Group 2, Test 4, Module 2 - Purge Time 96hr



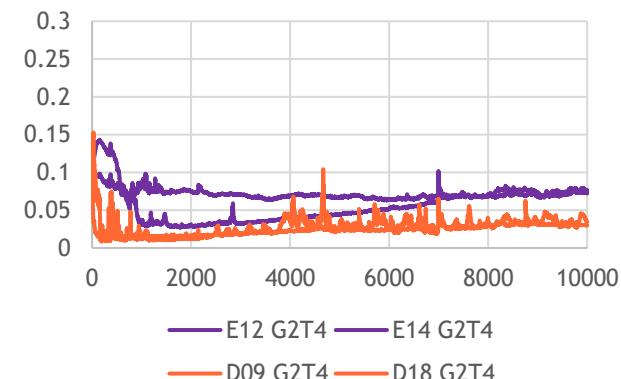
Group 1, Test 1, Module 2 - Purge Time 1hr



Group 2, Test 3, Module 1 - Purge Time 8hr

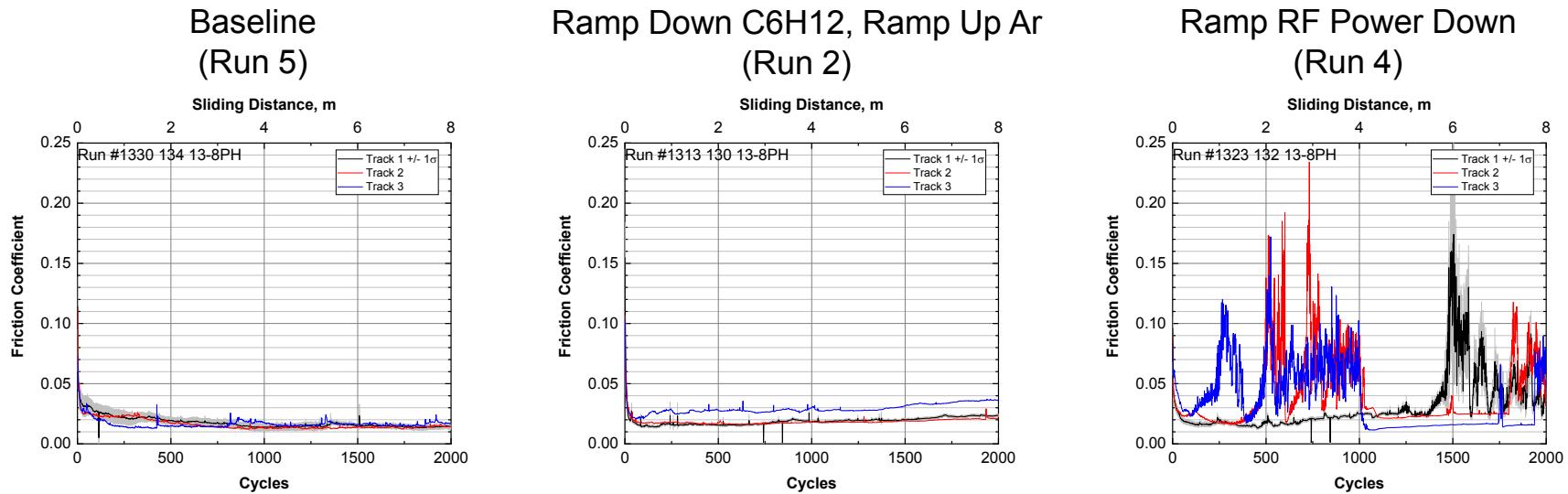


Group 2, Test 4, Module 1 - Purge Time 96hr



\* All labels are friction coefficient vs cycle  
\*\* 1 hour purges target <20 ppm O<sub>2</sub>/H<sub>2</sub>O

# Role of Surface Termination

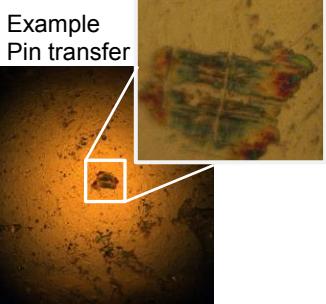


- All deposition runs same as baseline process, except for how the process ended
  - baseline involves shutting down gas precursors and RF power simultaneously
- The friction behavior was clearly modified by changing the shut-down process
  - there may be a decrease in run-in for some processes compared to baseline – more data (high throughput testing) will be used to evaluate significance

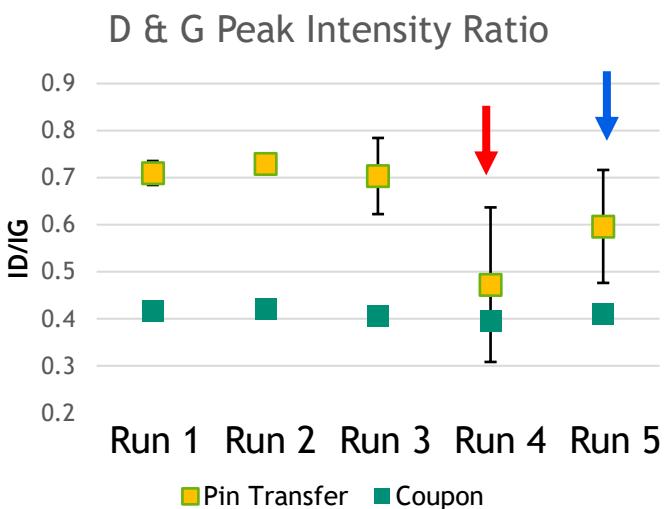
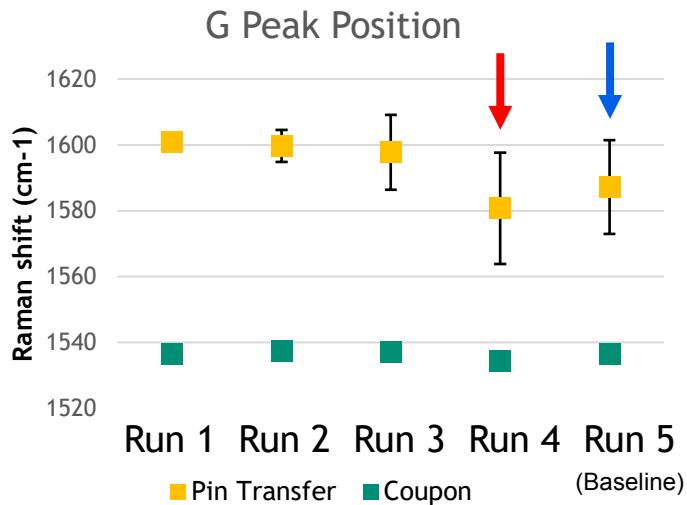
# Surface Termination - Raman



- As expected, spectra from coupons remained similar
- Spectra taken on pin transfer exhibited differences from baseline (Run 5) in all cases:
  - RF power ramp down (Run 4) – lower ID/IG and G Peak position; also exhibited highly erratic friction behavior
  - Runs 1,2,3 – higher ID/IG and G Peak position; exhibited similar friction behavior to baseline
- Results suggest that changes in friction behavior are linked to process changes that alter interfacial chemistry



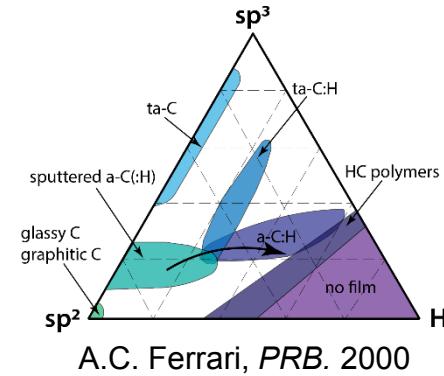
| Run ID | Process                            |
|--------|------------------------------------|
| 1      | Ar ion etch                        |
| 2      | Ramp Down $C_6H_{12}$ , Ramp Up Ar |
| 3      | Ramp Down $C_6H_{12}$              |
| 4      | Ramp RF Power Down                 |
| 5      | Baseline Process                   |



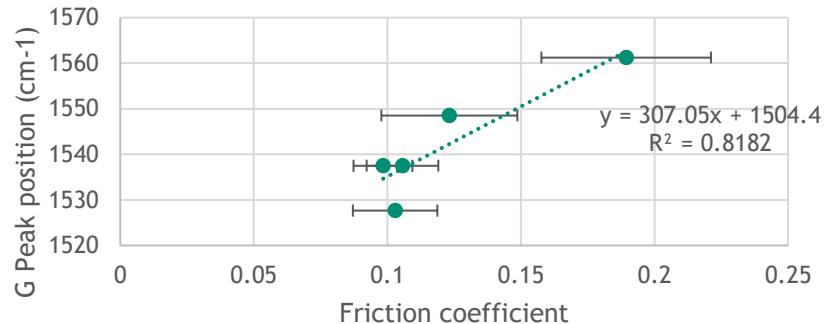
# Key Take-Aways



- Confirmed link between composition (via Raman/HFTF) and friction behavior for different vendors
- Surface termination simple route to change friction behavior
- Can utilize these relationships to design better coatings gear to run-in



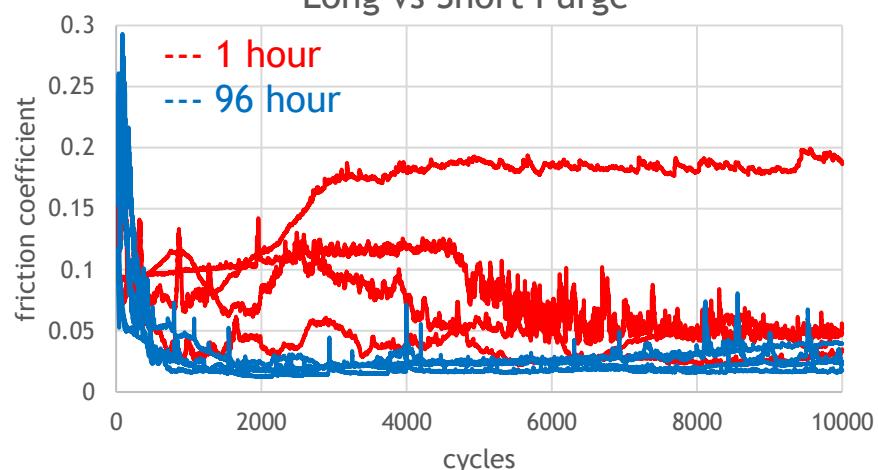
Initial Friction vs track G Peak Position



## Ongoing / Future Work

- Additional In situ testing (Raman/NEXAFS) for cycle resolved changes at surface
- What factors during purge change (ambient RGA)

Long vs Short Purge

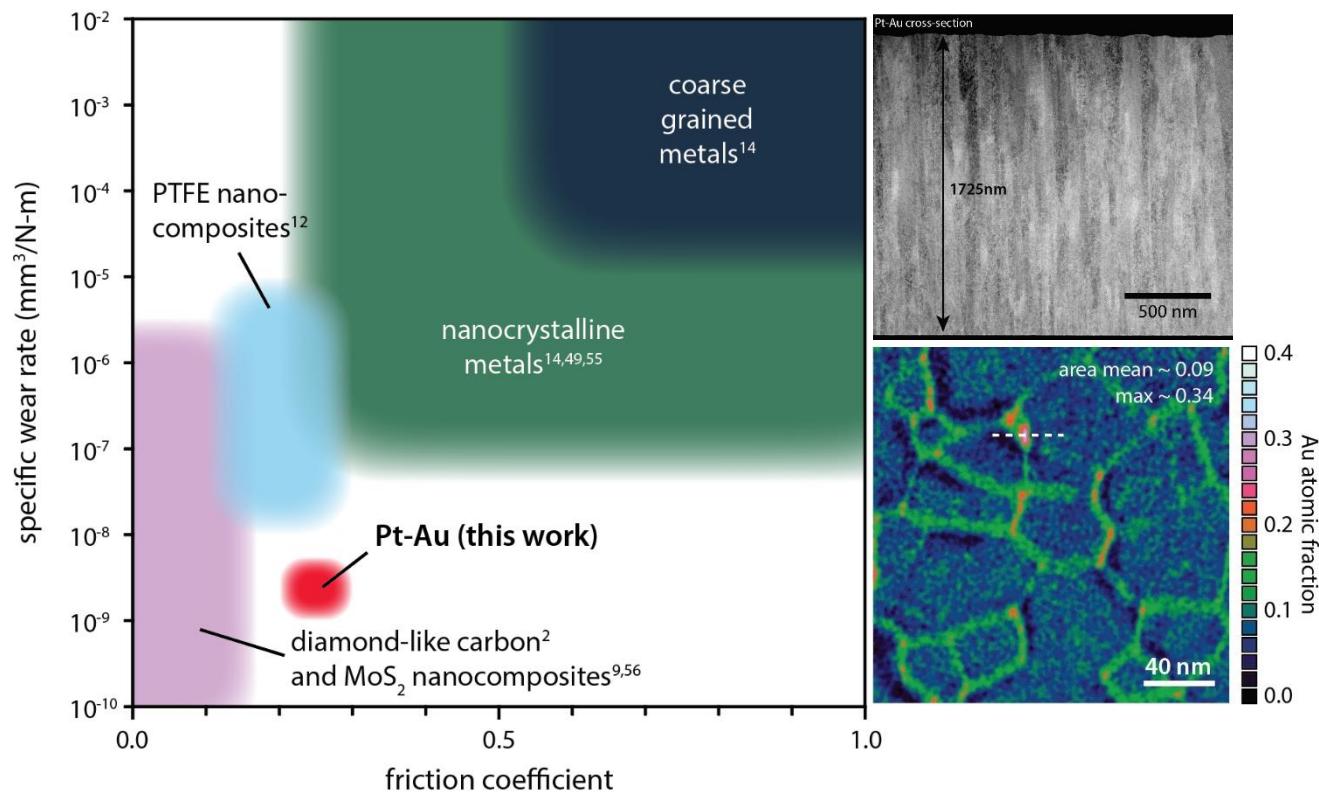




# Self-lubricating in situ carbon films (SLIC)

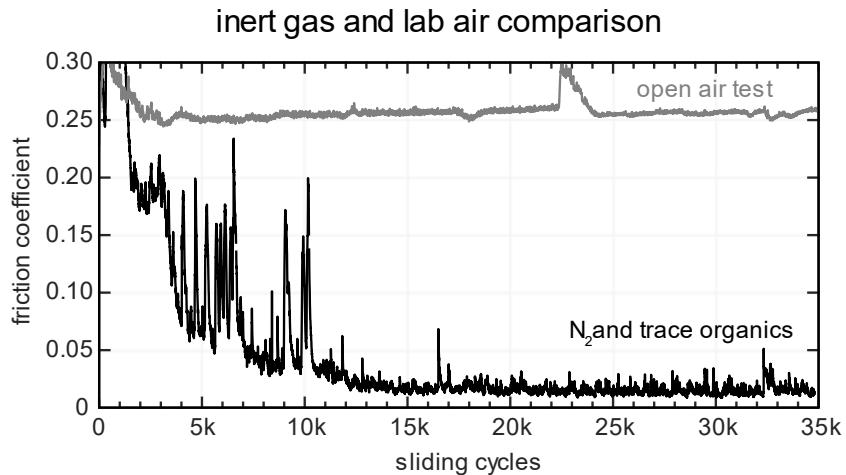


# Discovery: Ultra-Low Wear Pt-Au



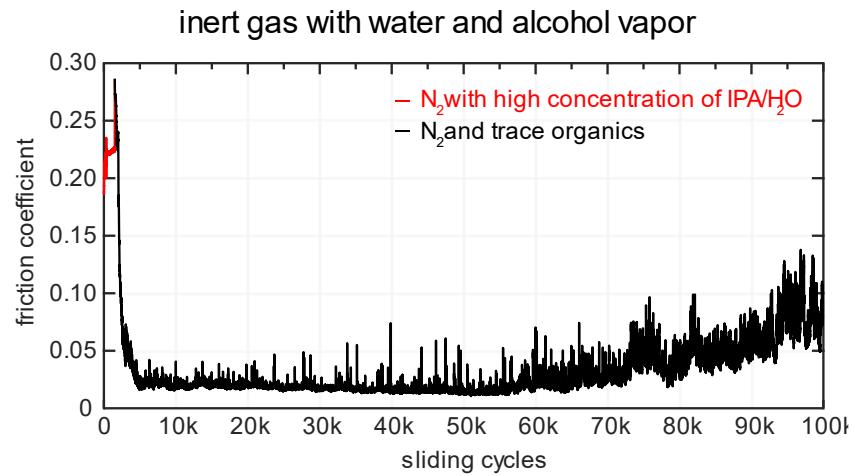
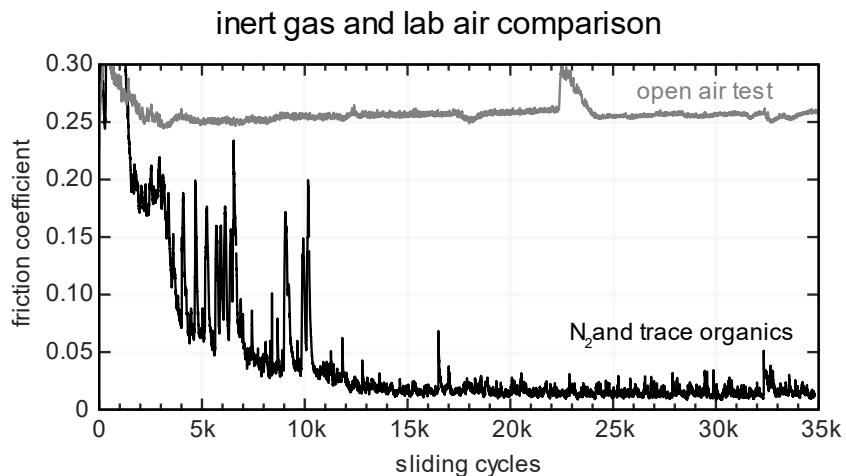
- Byproduct of LDRD on development of stable, ultra-nanocrystalline alloys
- Most tests run in air – needed to check performance in inert environments...

# Unexpected Tribocatalysis



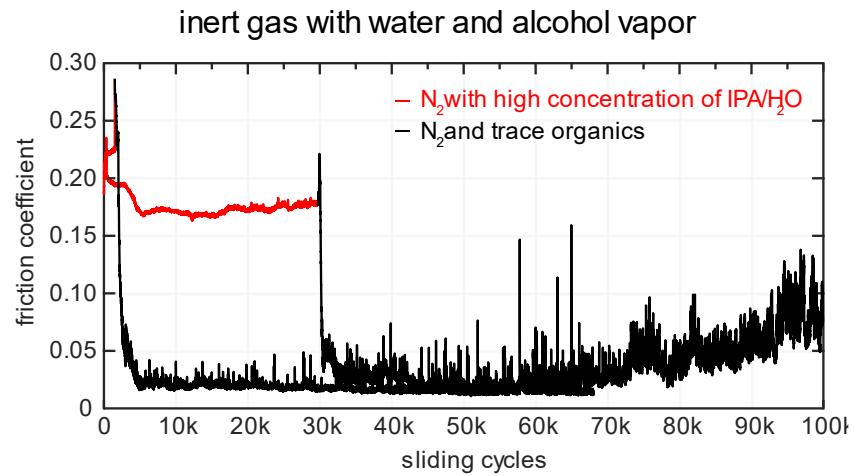
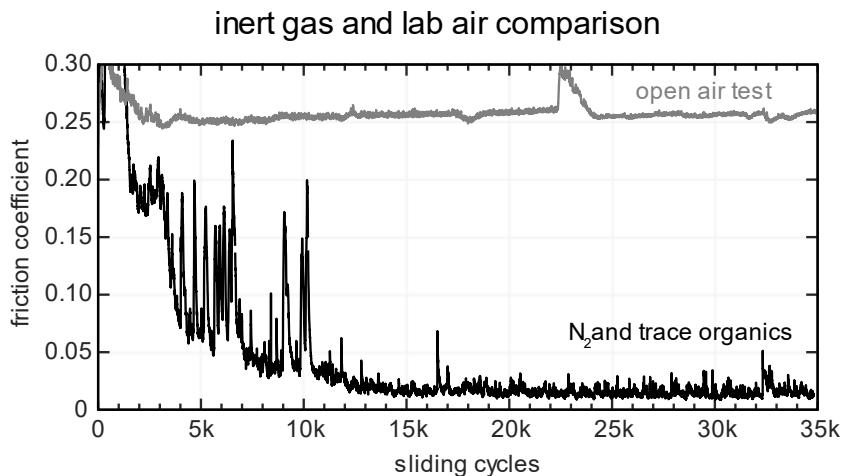
- Testing in inert environments lowers friction?

# Unexpected Tribocatalysis



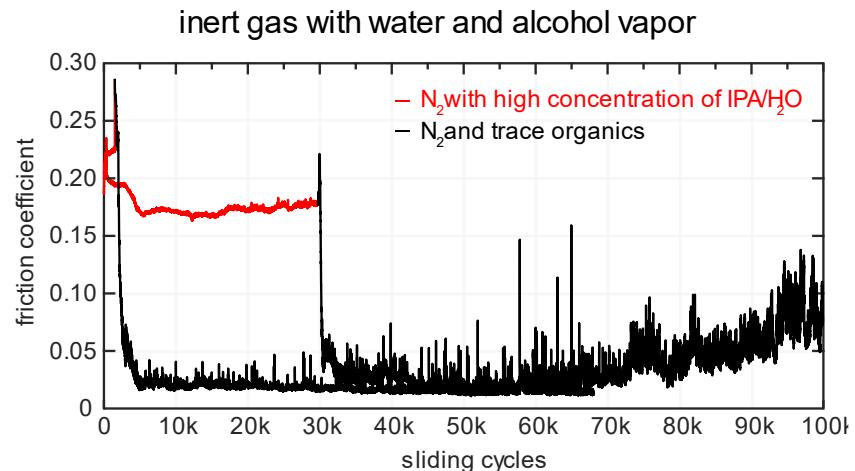
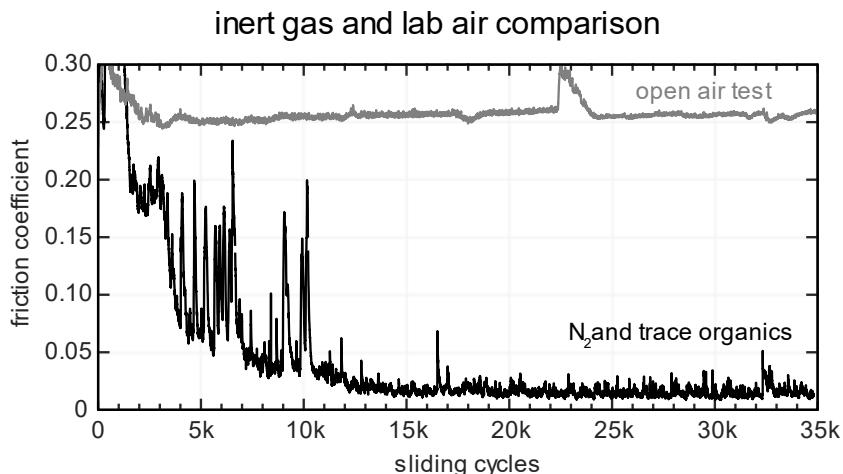
- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop...

# Unexpected Tribocatalysis

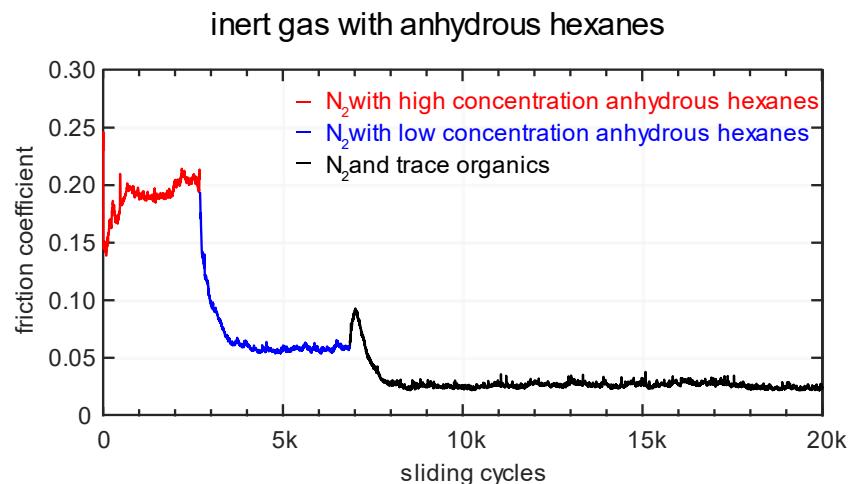


- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop... and prolongs it

# Unexpected Tribocatalysis



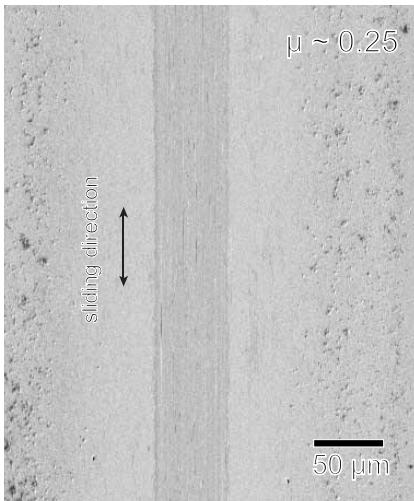
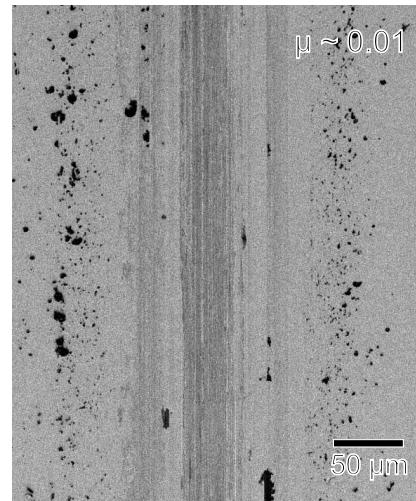
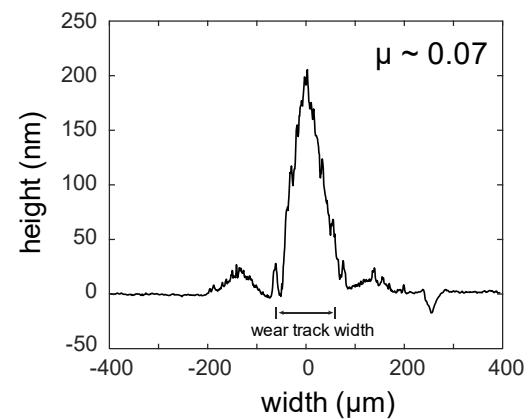
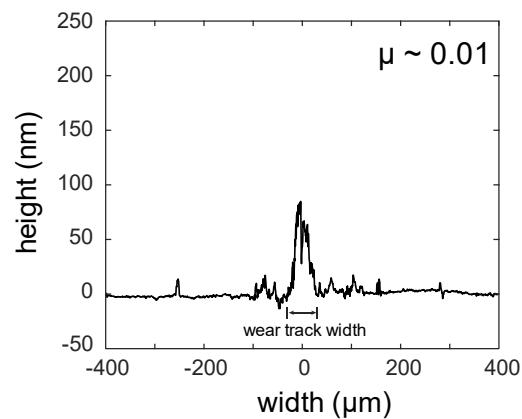
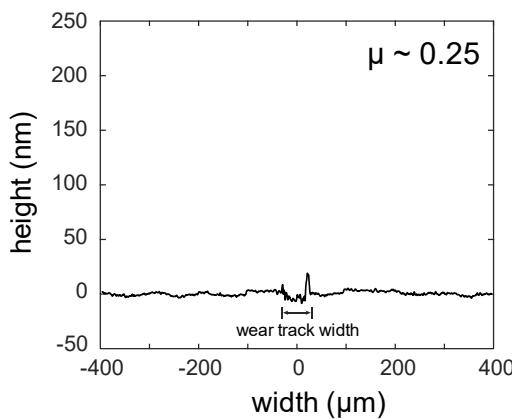
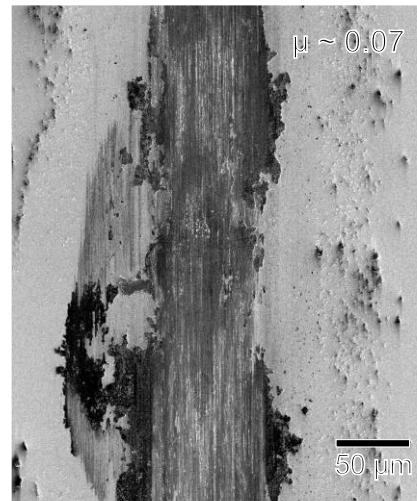
- Testing in inert environments lowers friction?
- Priming the enclosure with hydrated IPA accelerates drop... and prolongs it
- Any amount of anhydrous hexanes increased friction, with higher/lower friction at higher/lower concentrations
- Unclear what role water/oxygen play



# Accumulation is Key



A. lab air

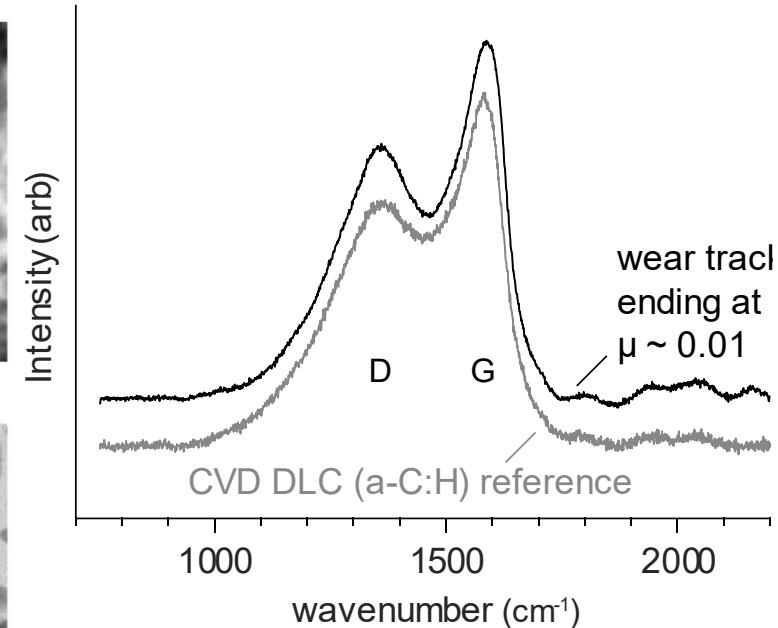
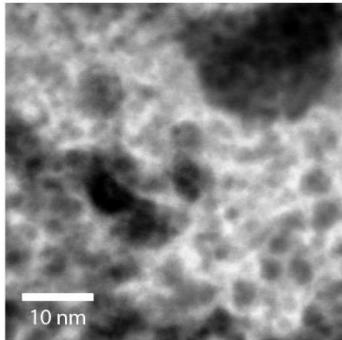
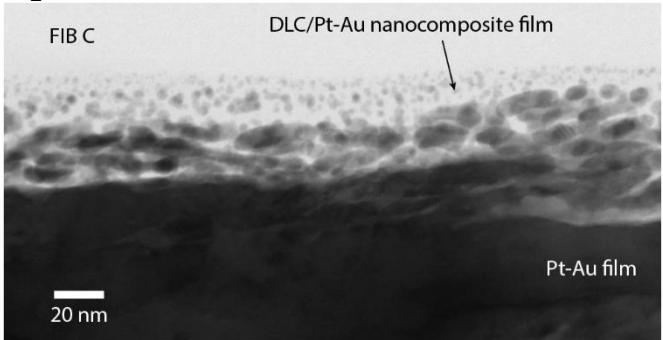
B.  $\text{N}_2$  and trace organicsC.  $\text{N}_2$  and high concentration IPA/H<sub>2</sub>O

- Concentration also affects film growth in wear scar
- Highest concentrations produces thick films unable to reach low friction state

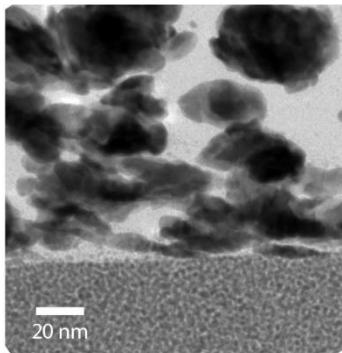
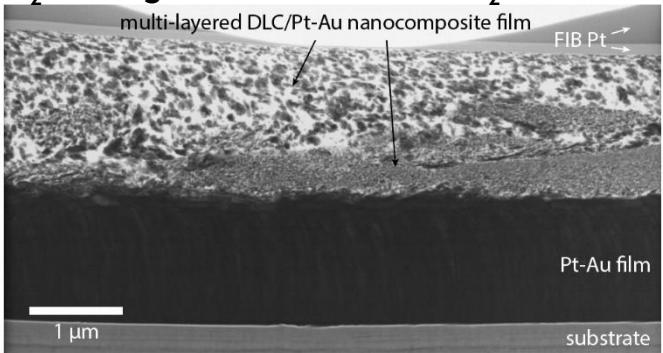
# A Tribo-polymeric Nanocomposite



$N_2$  and trace organics



$N_2$  and high concentration IPA/H<sub>2</sub>O

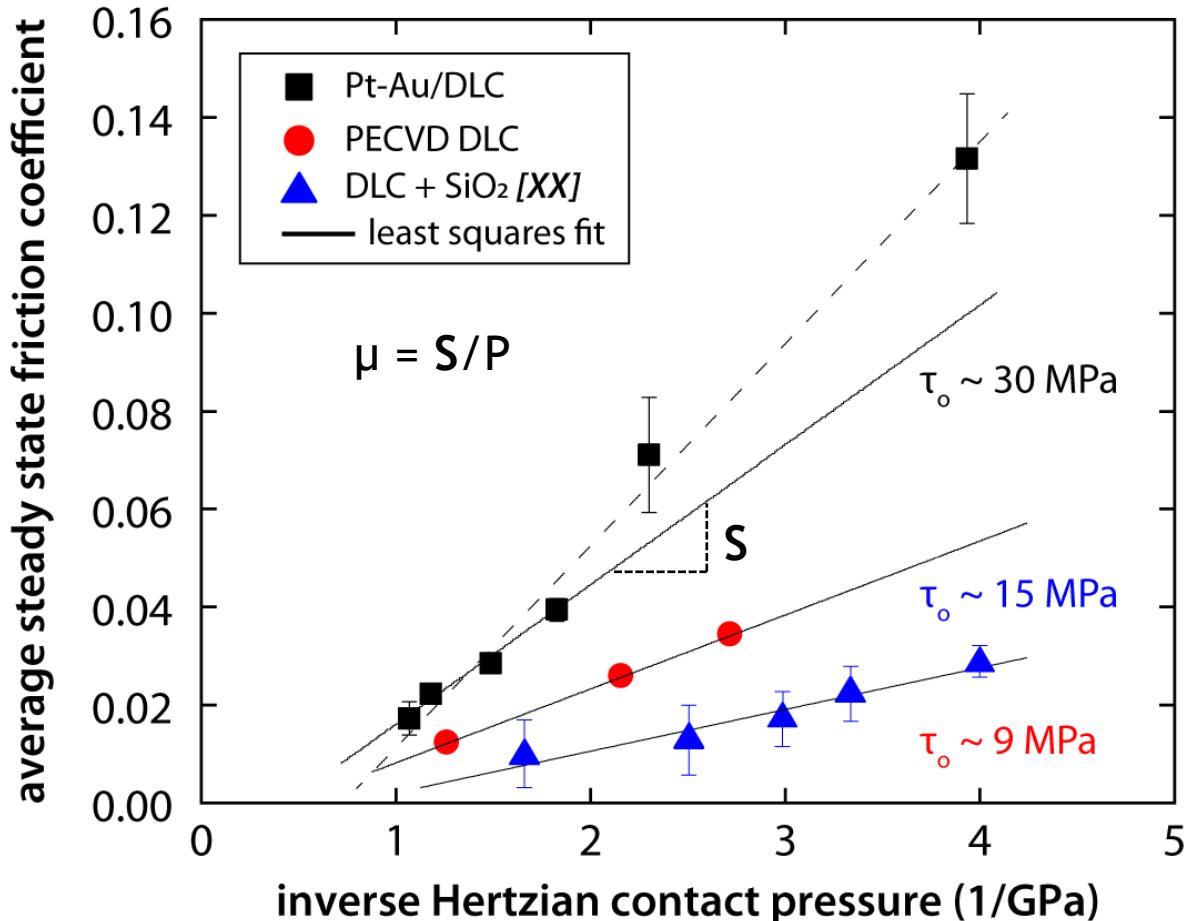


- Films are actually composite of Pt-Au nanoparticles and DLC-like carbon, confirmed by TEM & Raman
- High concentrations exhibit phases of larger, less mixed/layered particles, possibly limiting mixing & Pt interaction at surface

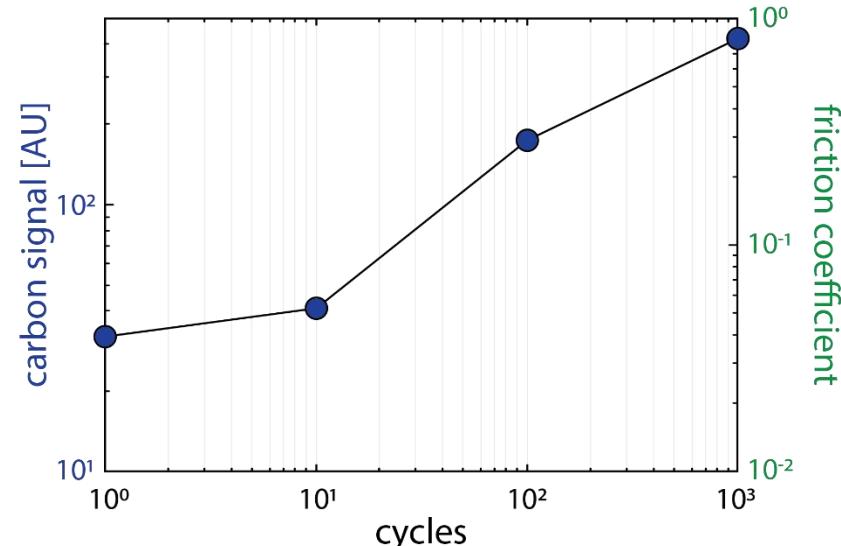
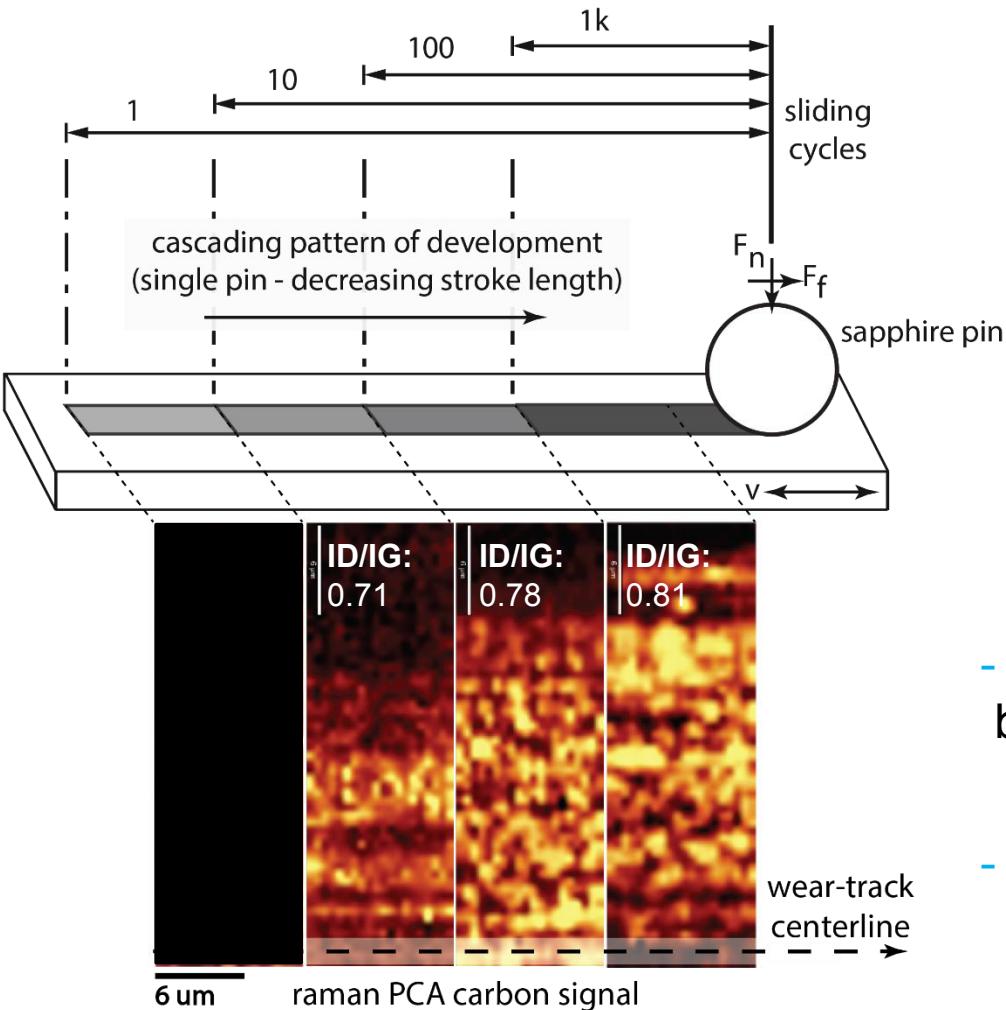
# Qualifying Shear Strength



- Can derive shear strengths from Hertzian contact model
- Shear strength comparable to commercially available
- Discrepancies may be due to composite nature of film and lower hydrogenation (20 vs 40%)

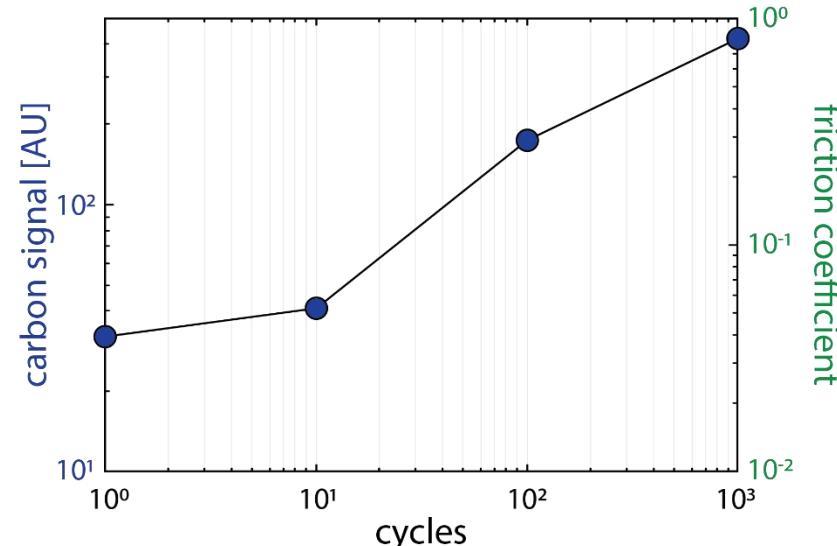
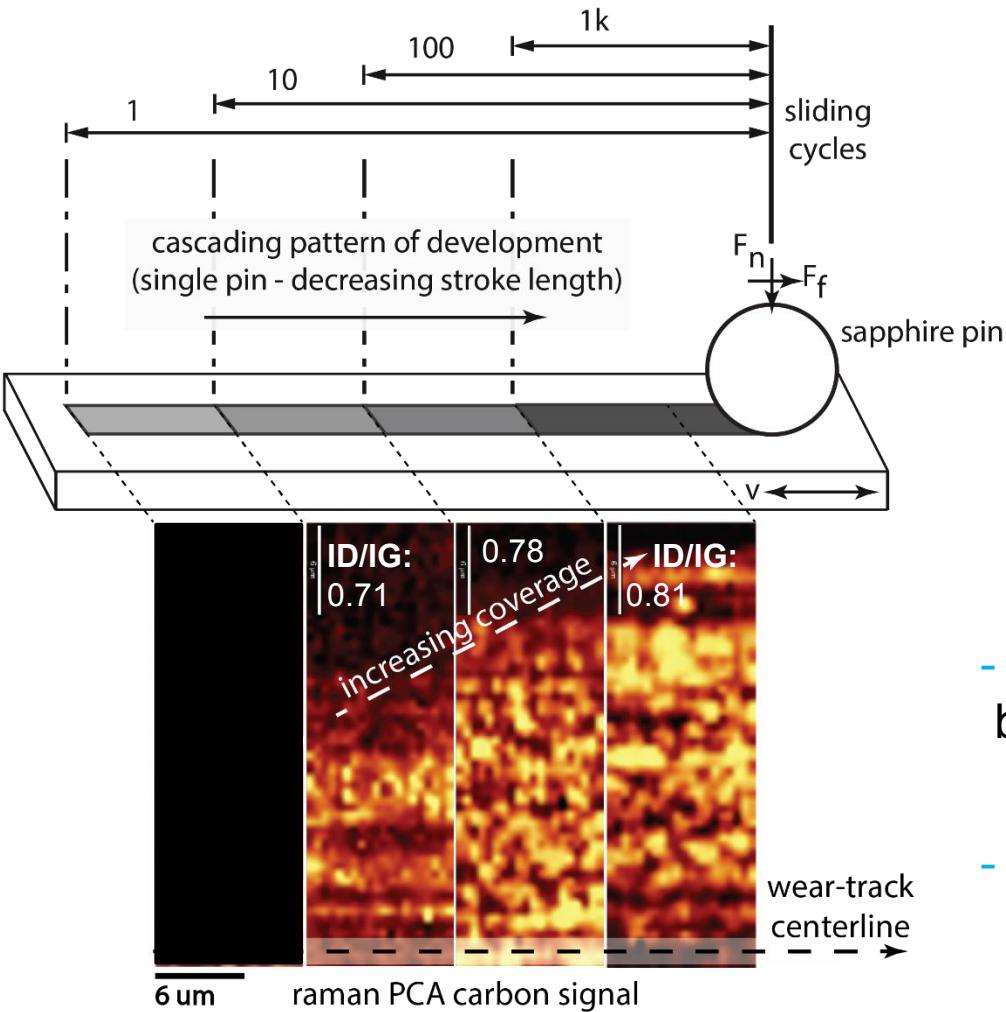


# Time Dependent Formation



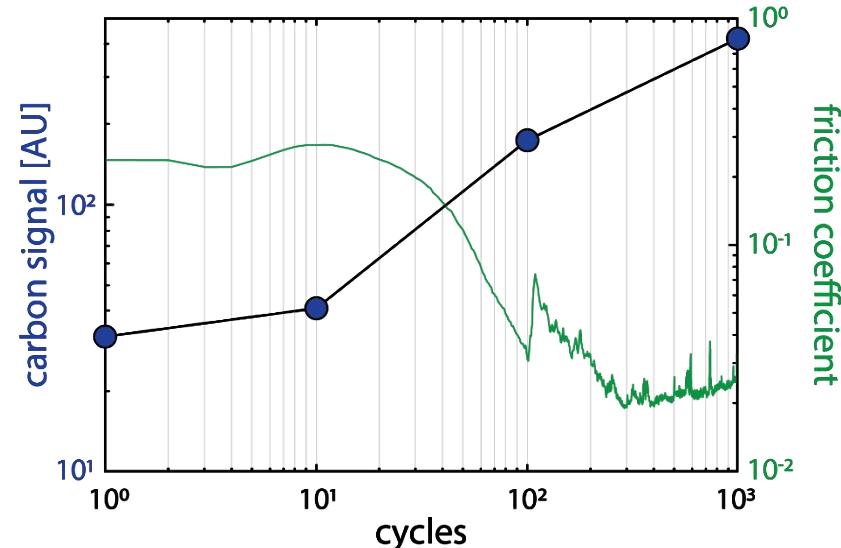
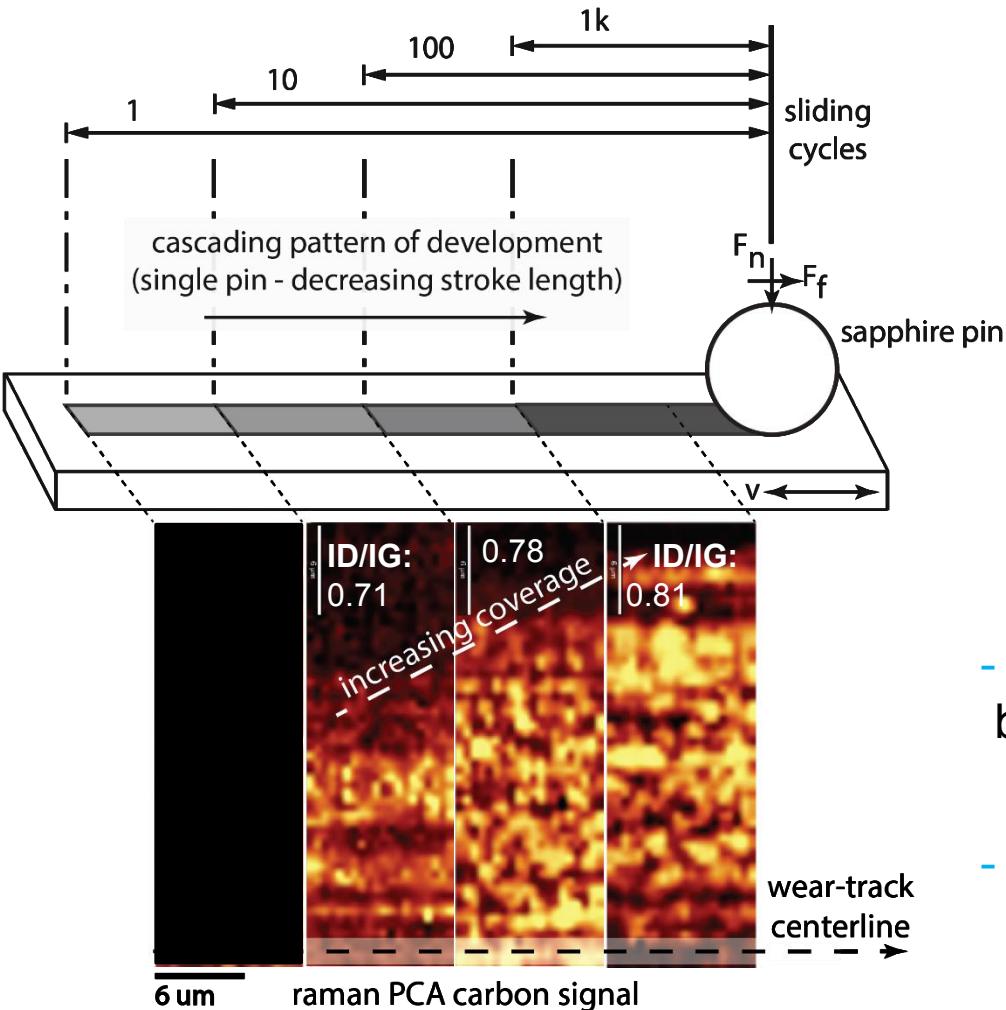
- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
  - Stronger carbon signals in wear track

# Time Dependent Formation



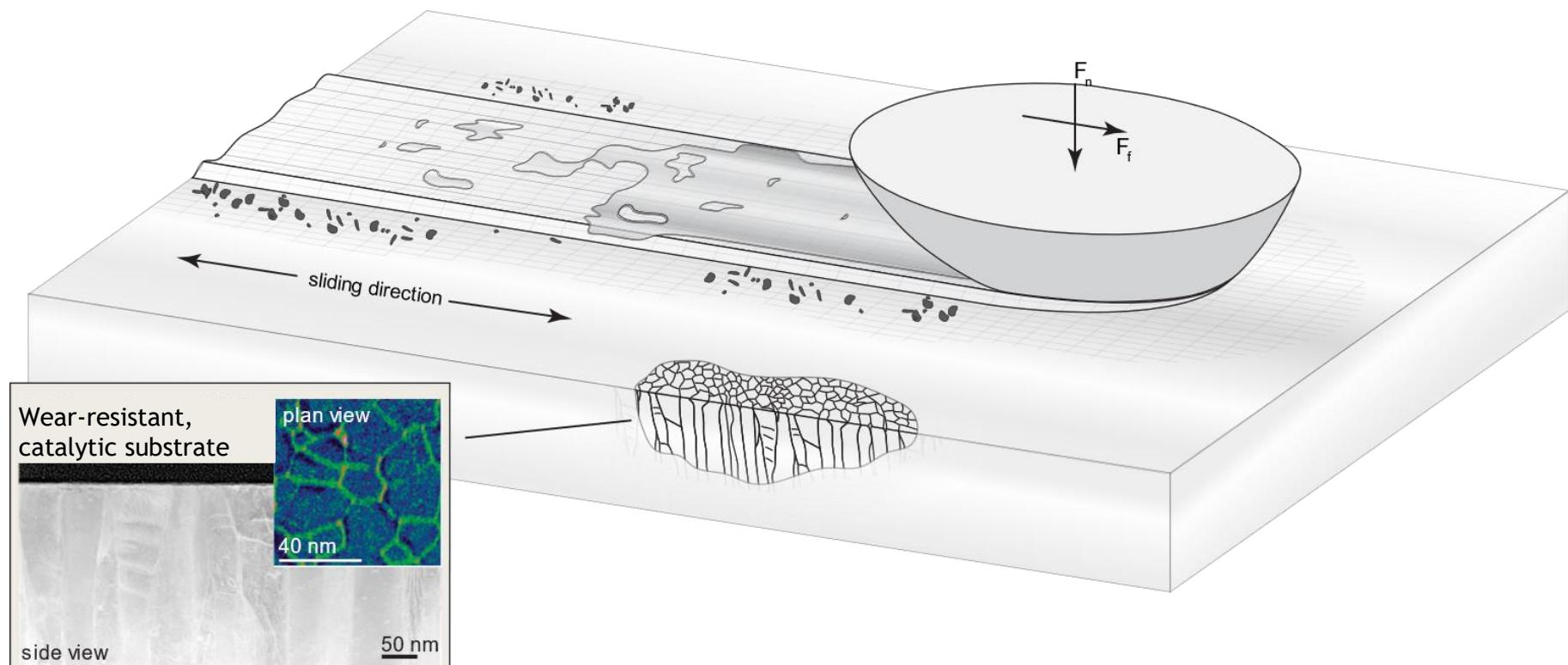
- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
  - Stronger carbon signals in wear track
  - Higher coverage in wear scar

# Time Dependent Formation

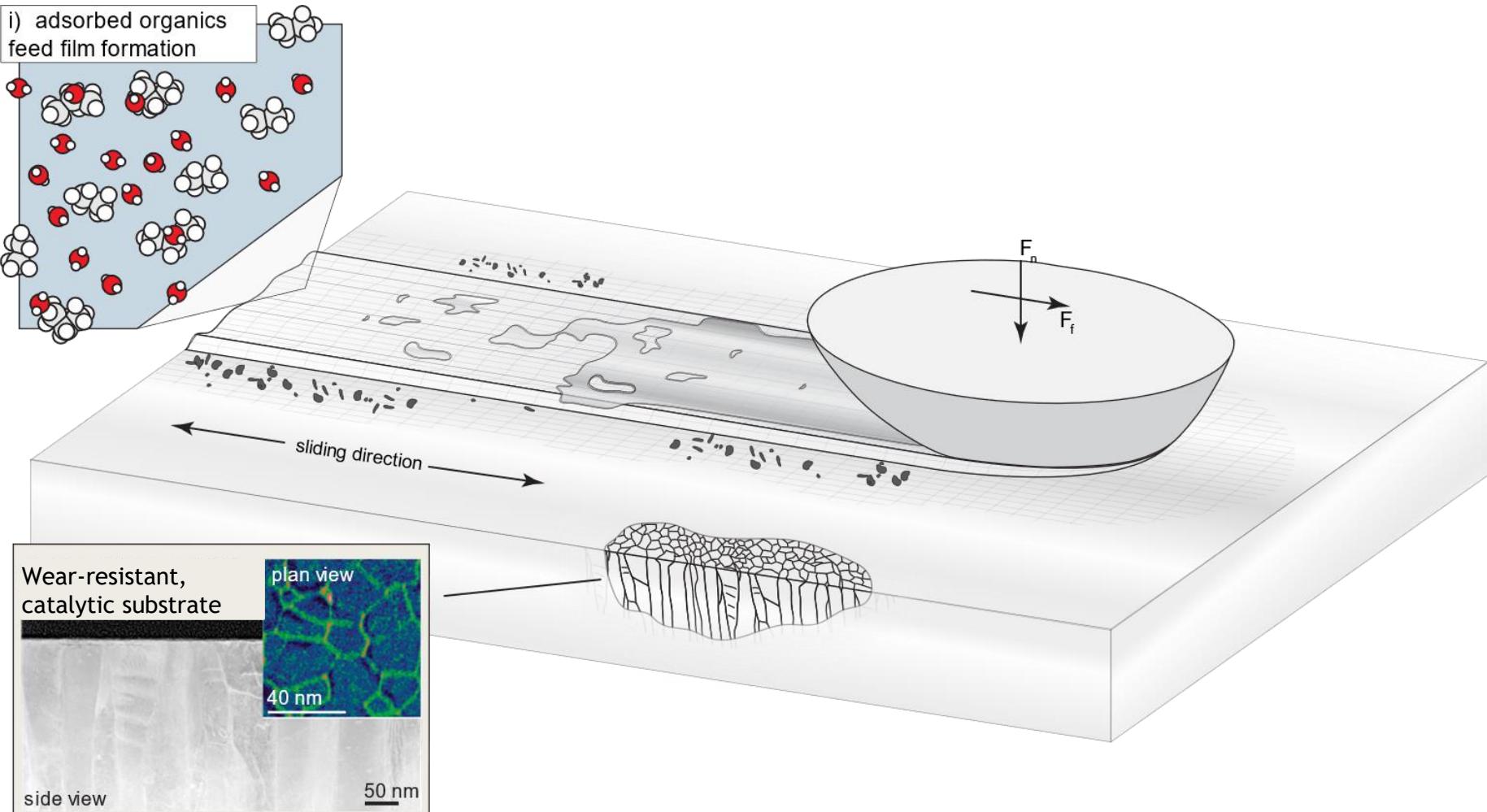


- Stripe tests elucidate time dependent behaviors
- Increasing cycle count leads to:
  - Stronger carbon signals in wear track
  - Higher coverage in wear scar
  - Decreasing friction coefficient

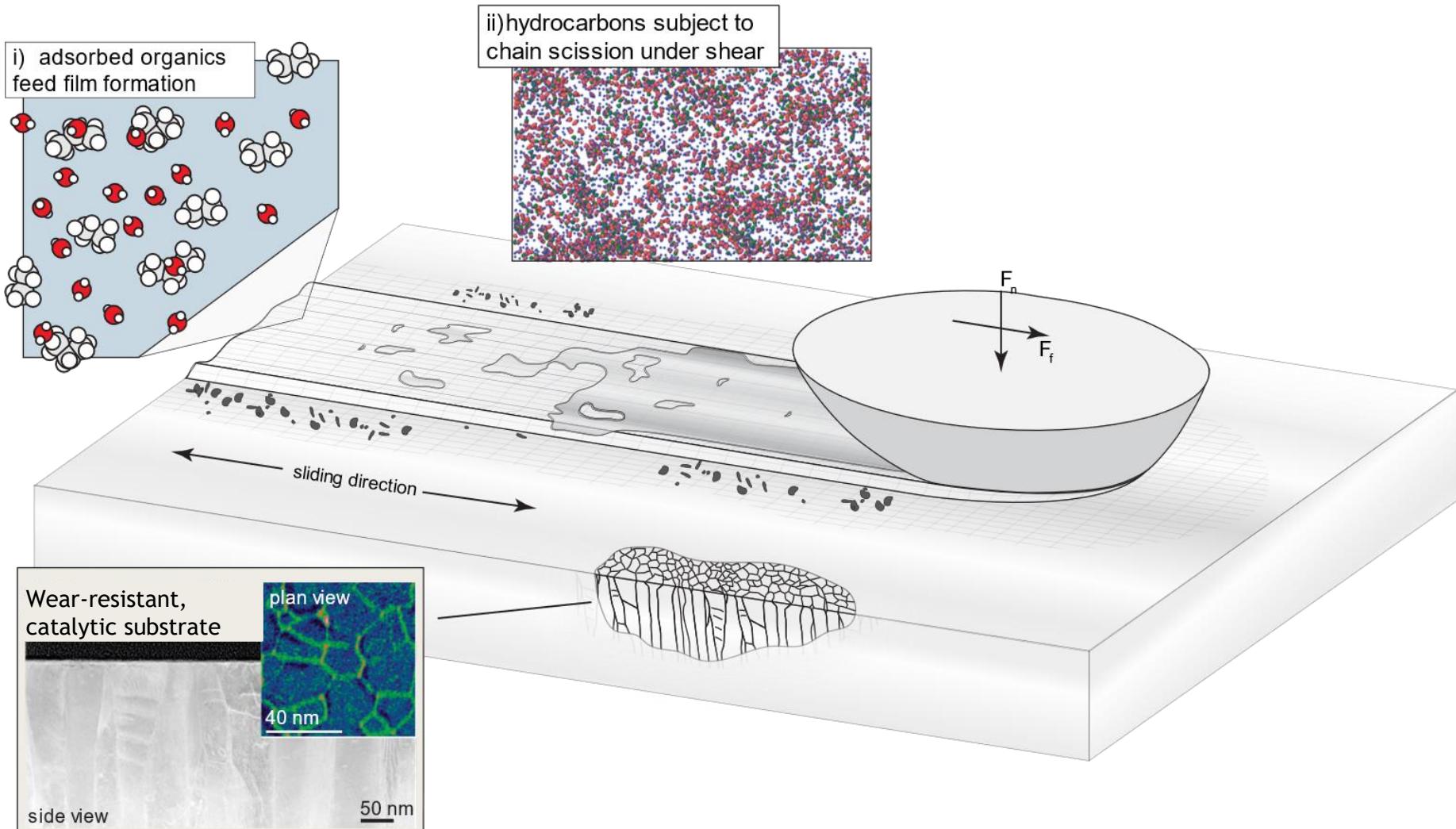
# Mechanisms of Formation



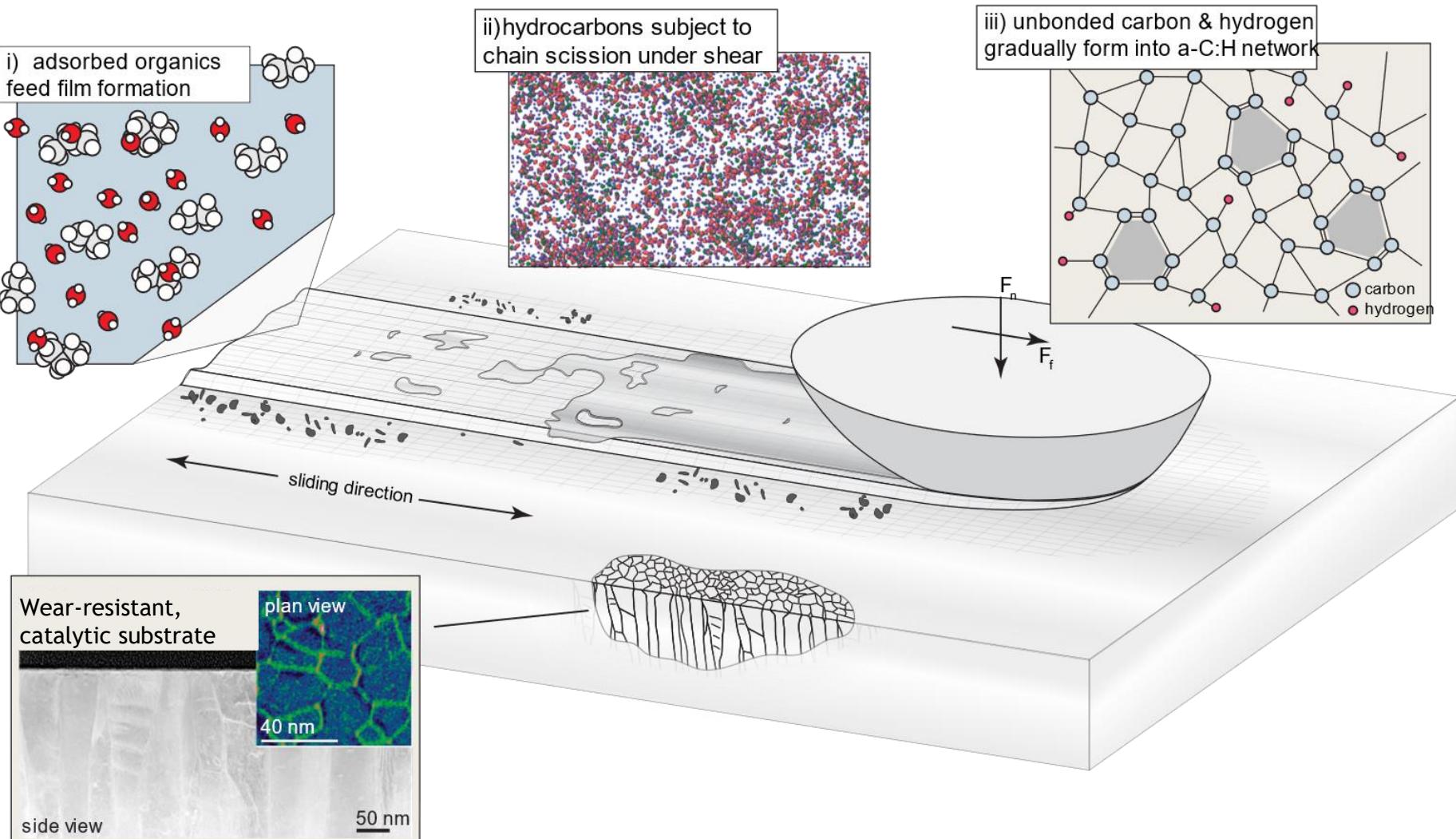
# Mechanisms of Formation



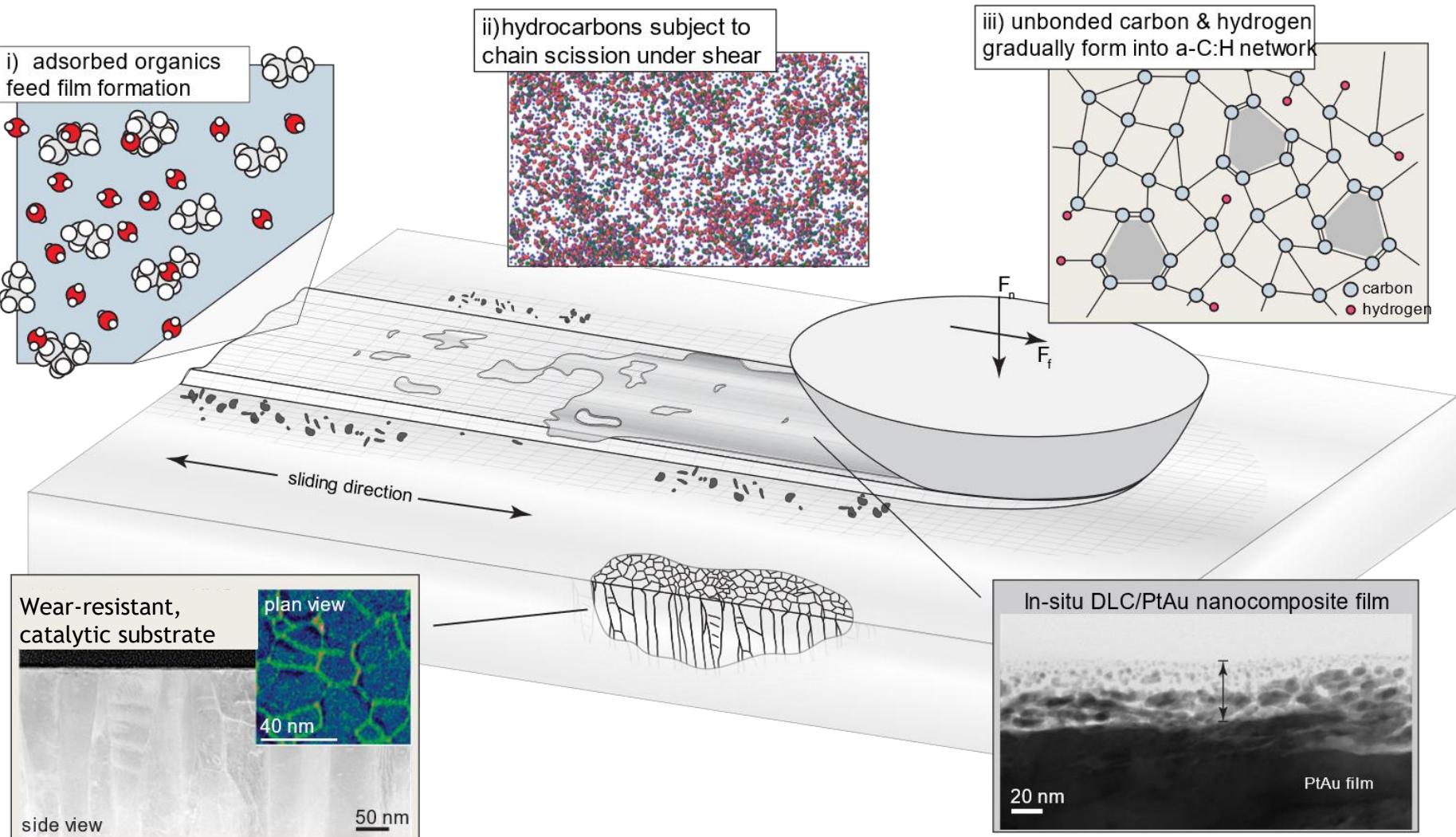
# Mechanisms of Formation



# Mechanisms of Formation



# Mechanisms of Formation



# Key Take-Aways

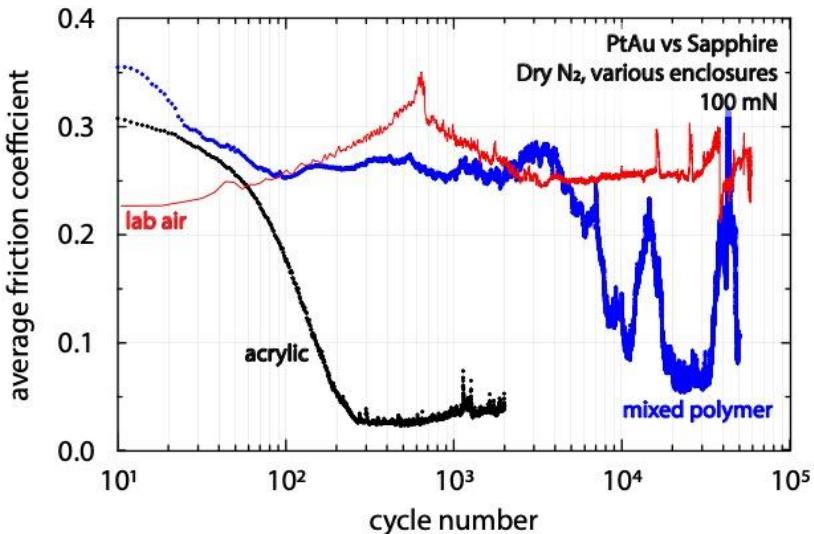
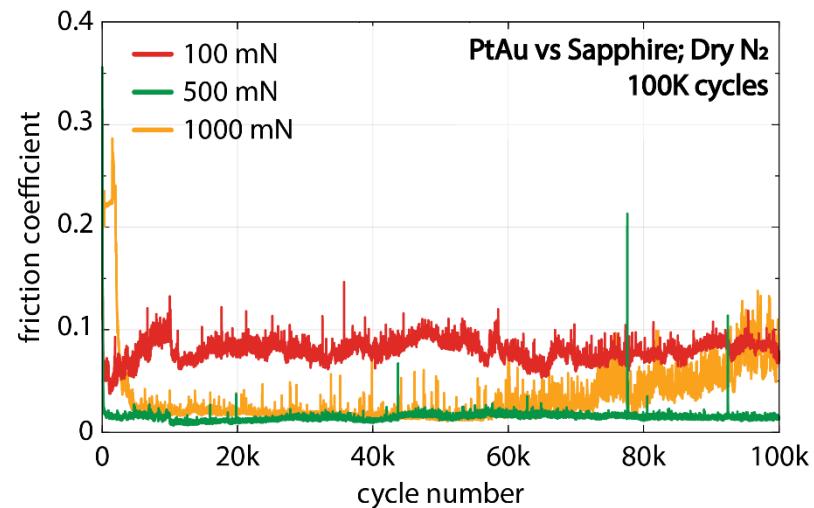


## Outcomes

- Trace concentrations of hydrocarbons enable formation of SLIC, a DLC-like tribofilm
- SLIC films exhibit stress/cycle dependent formation (& temperature in ethanol!)

## Ongoing / Future Work

- Understanding of underlying mechanisms behind competing stress, time & wear
- What do the resulting microstructures give us in terms of friction, wear or conductivity?
- What is the ideal composition or concentration of species in the environment for formation?
- Can we utilize as deposition method?



# Acknowledgements



## Sandia

- **Michael Dugger** MoS<sub>2</sub> Aging studies
- **Michael Chandross, Scott Bobbitt** MoS<sub>2</sub> MD/DFT
- **Frank DeIRio** AFM/KPFM on MoS<sub>2</sub>
- **Brendan Nation, JW Johnson, Morgan Jones, Brian Wisler** Experimental Support
- **Taisuke Ohta** XPEEM defect characterization on MoS<sub>2</sub>
- **Prof. Thomas Beechem** DLC characterization, in-situ Raman capability development



- **Florida State University Prof Brandon Krick, Tomas Babuska** in vacuo oxidation studies of MoS<sub>2</sub>

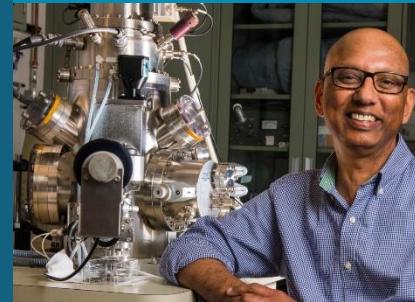
- **UT Austin Prof Filippo Mangolini, Robert Chrostowski** NEXAFS/APPES experiments on MoS<sub>2</sub>/DLC





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National  
Laboratories

# Thank You



# Questions?

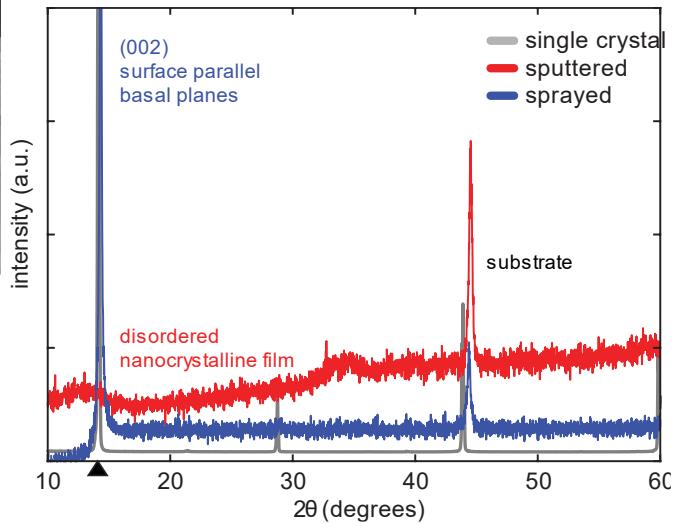
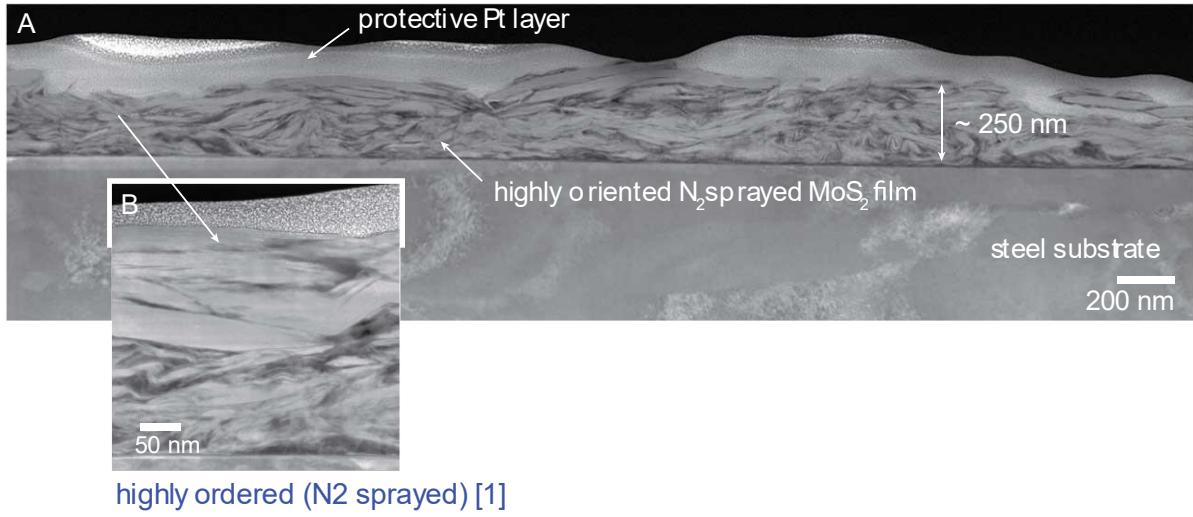


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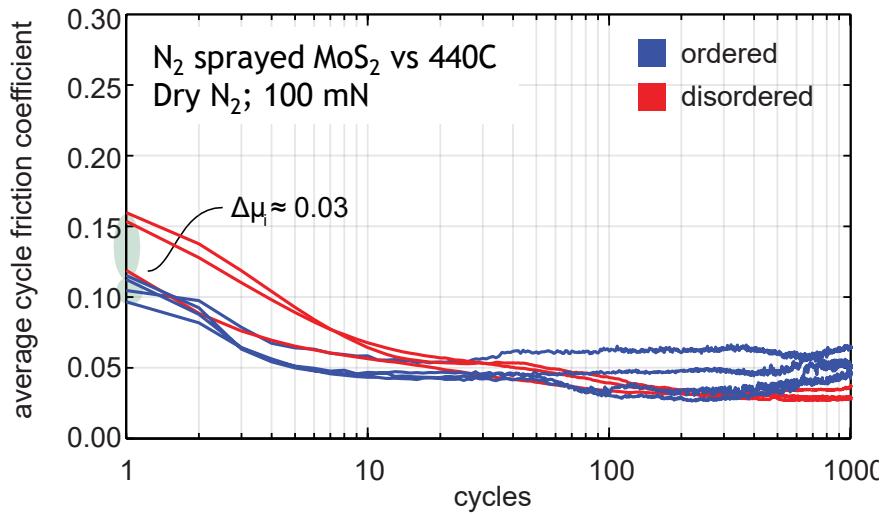
## APPENDIX



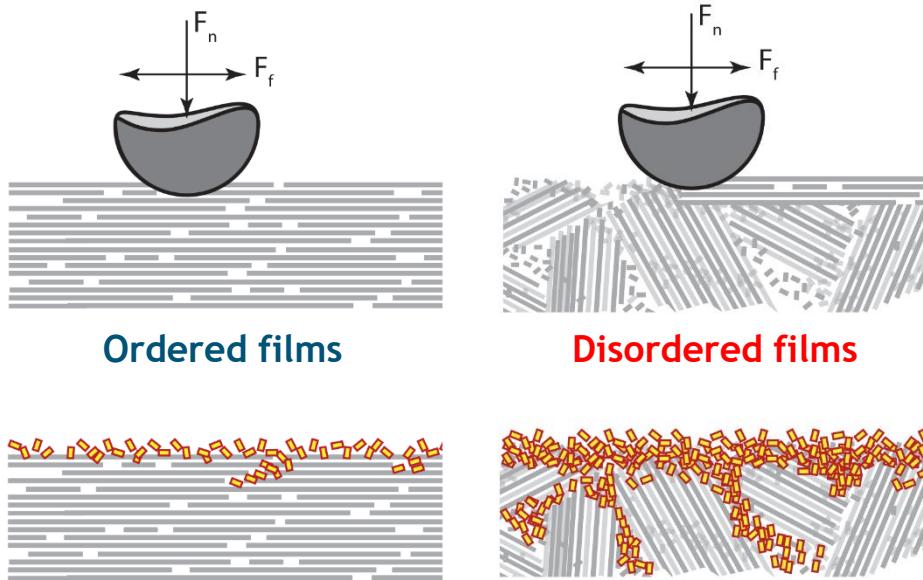
# Solution? Start Simple



- To make things simple, we focused on ordered, impinged films
  - Blast  $N_2/MoS_2$  onto surface to get films close to basally oriented as deposited
  - Exhibit lower initial friction coefficients
  - Ordering may help prevent degradation

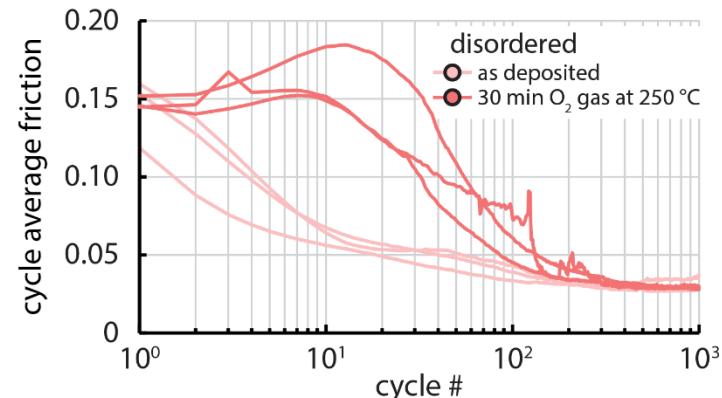
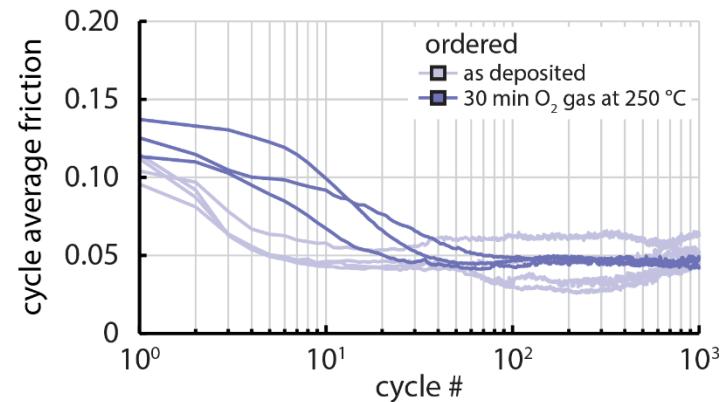


# Accelerated Aging Study

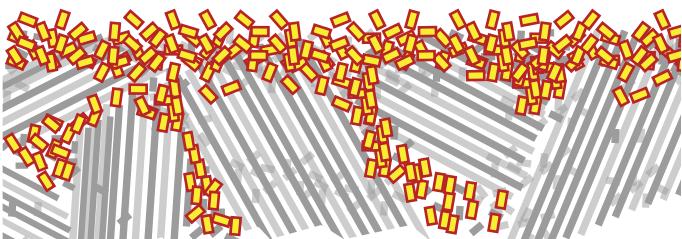


- Less defects/edges may also limit oxidation to surface
- Exposures of 30 min  $O_2$  @ 250 °C (also AO) show ordered films exhibit less oxidation & surface limited, reducing effects on run-in

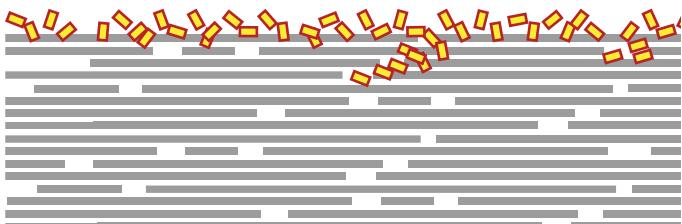
First demonstration of microstructure's role in limiting oxidation; run-in (ordered) surfaces help prevent aging related issues



# Run-In Factor: Oxidation



disordered structure



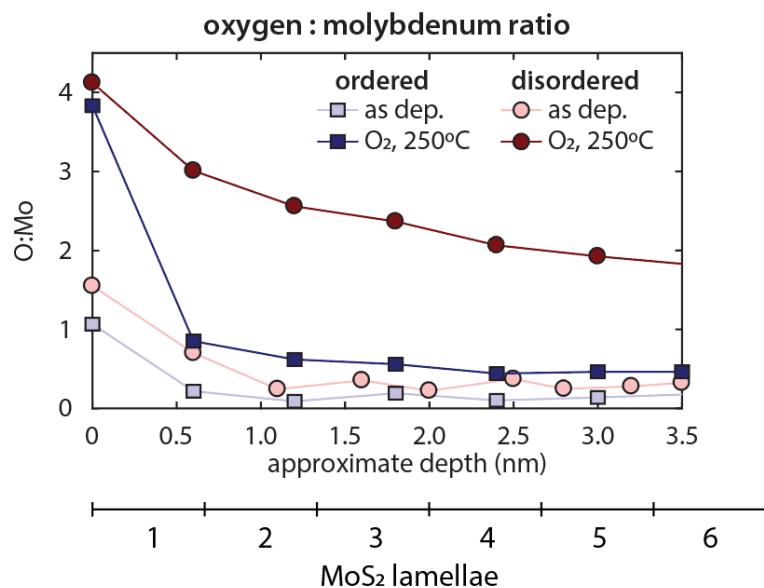
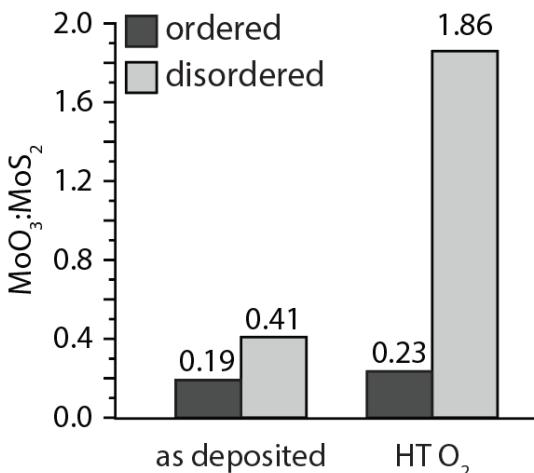
highly-ordered structure

- Oxidation resistance should benefit in the same way that run-in does from ordered surfaces
- Higher degree of basal orientation and less available edge site (large crystals) should reduce oxidation
- Ordered structure also provides more tortuous path into the bulk for further interactions

# Oxidation vs Microstructure – XPS & LEIS Study



Mo 3p signal - MoO<sub>3</sub>:MoS<sub>2</sub> ratio



- Look at amount of Mo as sulfide or oxide after exposures to O<sub>2</sub> @ 250C and Atomic Oxygen (30 min)

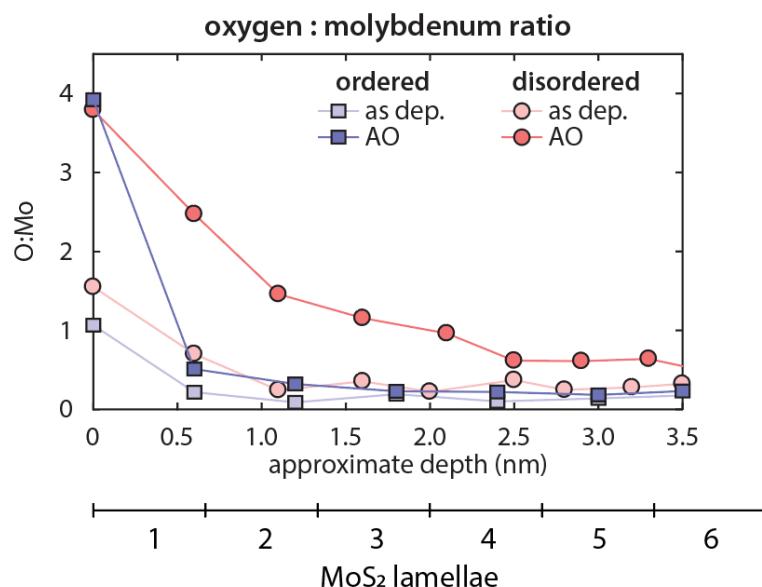
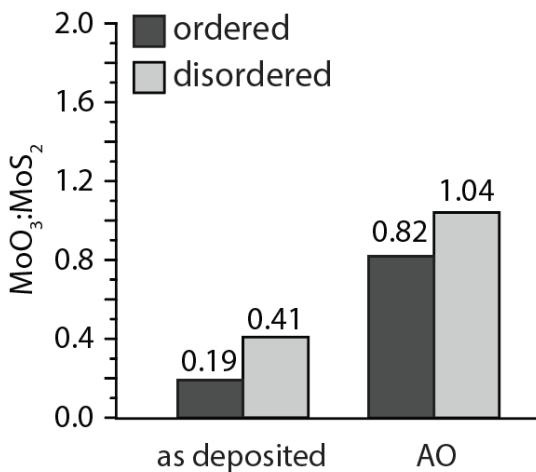
## Oxygen Gas (30 min @ 250°C)

- XPS indicates minimally more oxide for ordered films while disordered films have more
- LEIS shows this is mostly surface limited for ordered films and through the surface for disordered

# Oxidation vs Microstructure – XPS & LEIS Study



Mo 3p signal -  $\text{MoO}_3:\text{MoS}_2$  ratio



- Look at amount of Mo as sulfide or oxide after exposures to O<sub>2</sub> @ 250C and Atomic Oxygen (30 min)

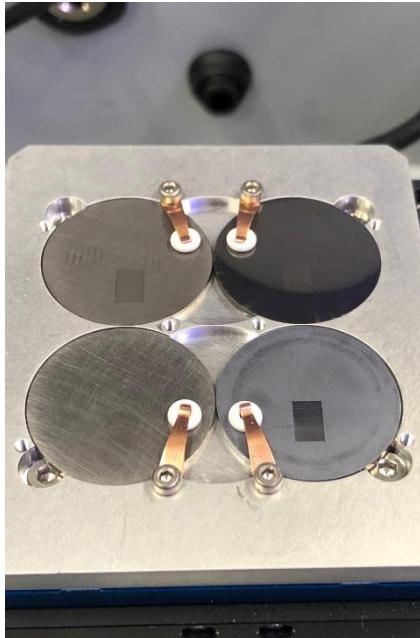
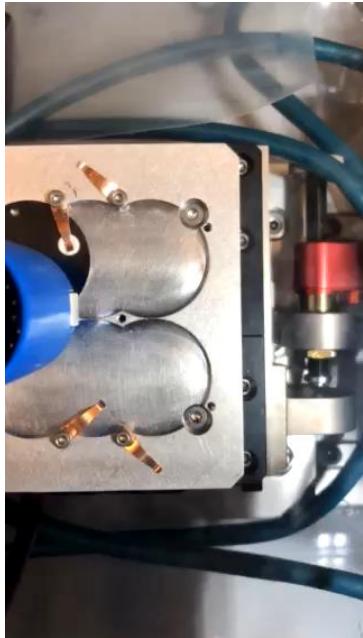
## Oxygen Gas (30 min @ 250°C)

- XPS indicates minimally more oxide for ordered films while disordered films have more
- LEIS shows this is mostly surface limited for ordered films and through the surface for disordered

## Atomic Oxygen (30 min @ RT)

- AO exposures show similar increases in oxidation via XPS
- Again LEIS shows oxygen only at surface for ordered films and not much below the surface for disordered

- Perform XPS inside versus outside rubbed area
- Return to run-in area after aging for additional friction testing



# Link #4: Structurally driven temperature dependent friction of $\text{MoS}_2$ - friction

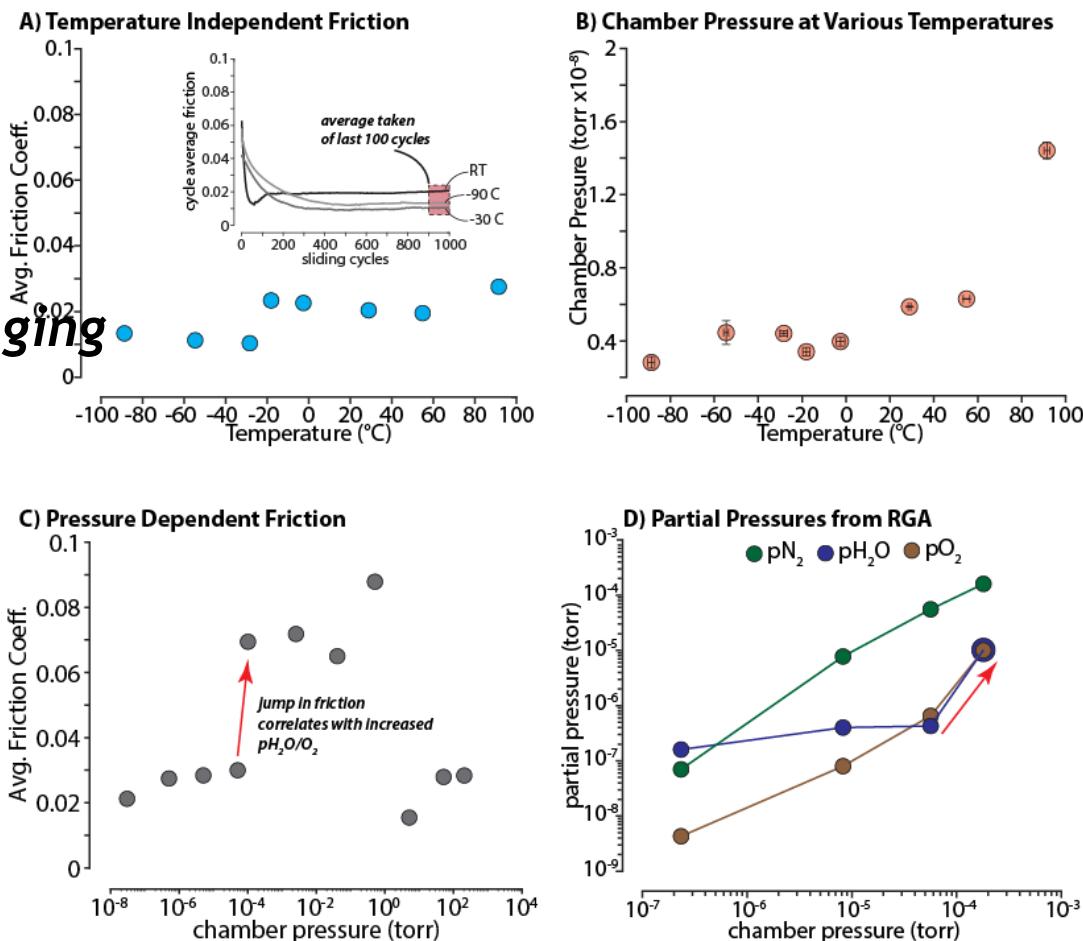


## Constant Pressure, Changing Temperature

- In UHV, at constant pressure, the friction behavior of  $\text{MoS}_2$  is INDEPENDENT of temperature
- Coating shows no measurable wear at any temperature

## Constant Temperature, Changing Pressure

- In UHV, at constant temperature, the friction behavior of  $\text{MoS}_2$  changes with pressures
- A high friction regime at moderate vacuum ranges
- Friction drops again at higher pressures
- The increase in friction correlates with an increase in the  $\text{pH}_2\text{O}$  and  $\text{pO}_2$

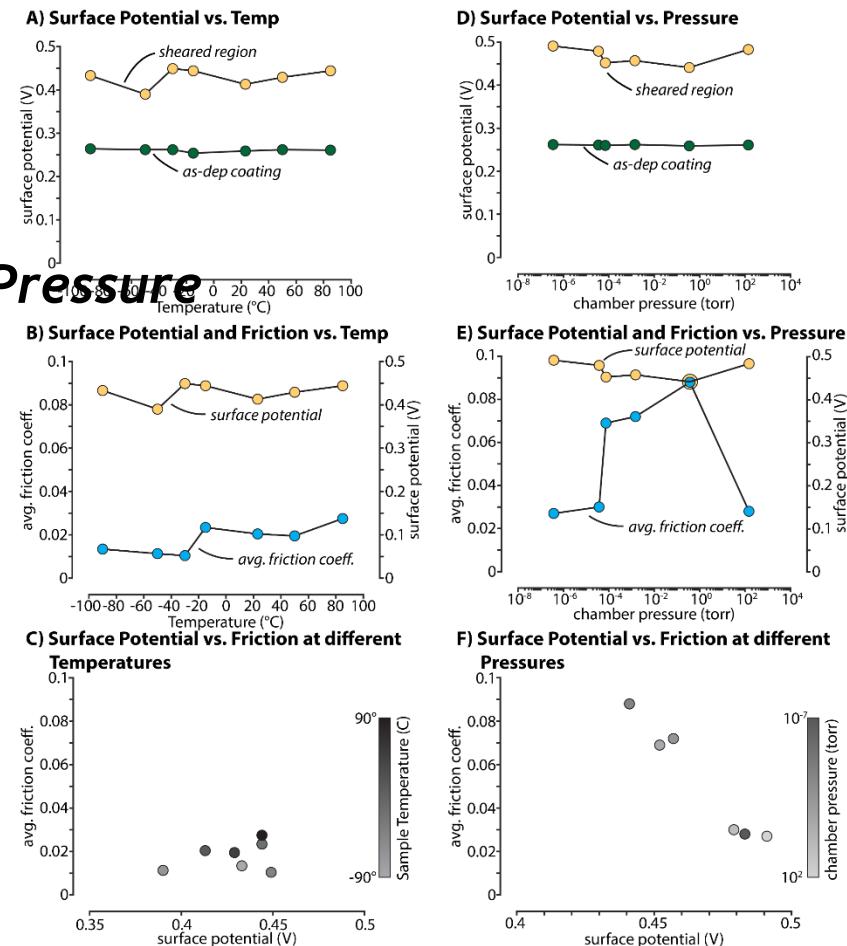


**Question: How is structure driven by temperature and pressure?**



## Constant Pressure, Changing Temperature

- Sheared regions at all temperatures have higher surface potential than coating
- No apparent trend between **Constant Temperature, Changing Pressure** friction and surface
- ~~As temperature increase in surface potential = increase in friction coefficient~~
- Increase in surface potential = decrease in friction coefficient
- Strong relationship between friction and surface potential



**Takeaway: Structure does not evolve with temperature, only pressure**