



# Minority Carrier Lifetime and Recombination Dynamics in Strain-Balanced InGaAs/InAsSb Superlattices

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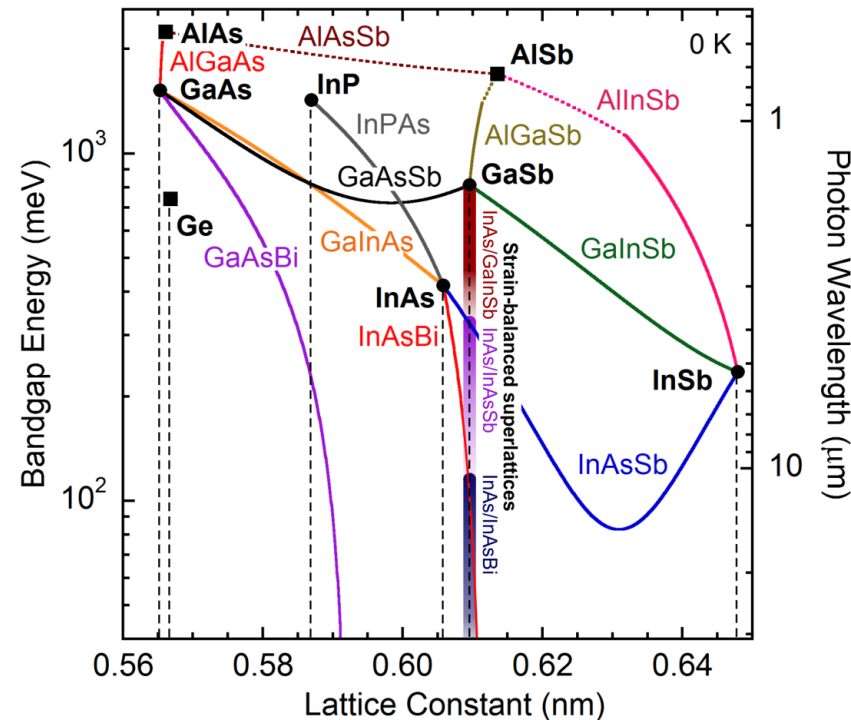
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# Infrared Materials with Lattice-Matched Substrates

III		IV	V
5 <b>B</b> Boron 10.81	6 <b>C</b> Carbon 12.011	7 <b>N</b> Nitrogen 14.007	
13 <b>Al</b> Aluminium 26.981...	14 <b>Si</b> Silicon 28.085	15 <b>P</b> Phosphorus 30.973...	
31 <b>Ga</b> Gallium 69.723	32 <b>Ge</b> Germanium 72.63	33 <b>As</b> Arsenic 74.92160	
49 <b>In</b> Indium 114.818	50 <b>Sn</b> Tin 118.710	51 <b>Sb</b> Antimony 121.760	
81 <b>Tl</b> Thallium 204.38	82 <b>Pb</b> Lead 207.2	83 <b>Bi</b> Bismuth 208.98...	



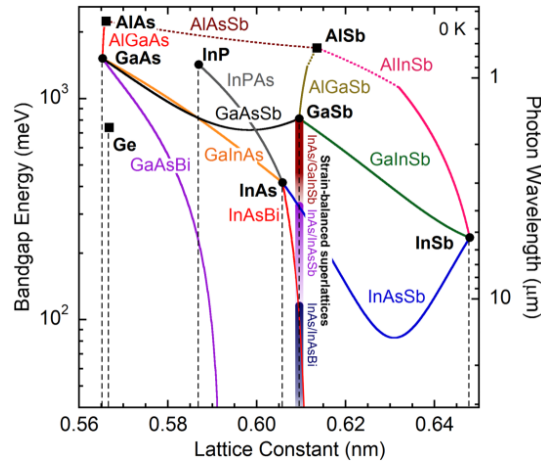
Strain-balanced or lattice-matched superlattices can be grown arbitrarily thick on conventional III-V substrates

Reducing the effective bandgap of the superlattice comes at the expense of reduced absorption (wavefunction overlap) shown by color gradient

The ground state absorption coefficient can be maximized by optimizing the designs for maximum wavefunction overlap

Design freedom afforded by additional alloy constituents allows for simultaneous optimization of band edge alignment and strain profile

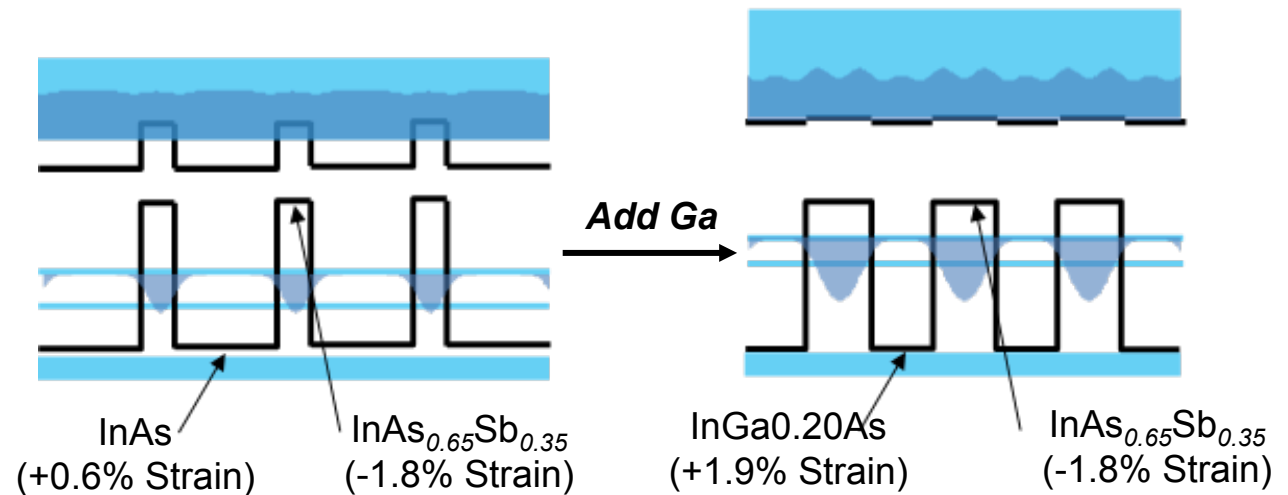
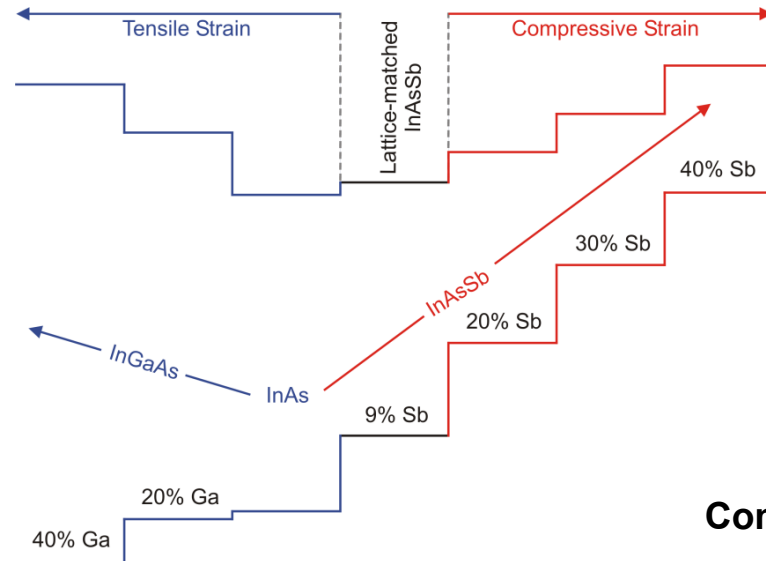
# Impact of Adding Ga in InGaAs/InAsSb Superlattices



In InAs/InAsSb, the lightly tensile InAs layers must be thicker than the more heavily compressive InAsSb layers to balance the strain

Incorporating Ga into InAs yields increasingly more tensile InGaAs, resulting in a more symmetric strain-balance profile

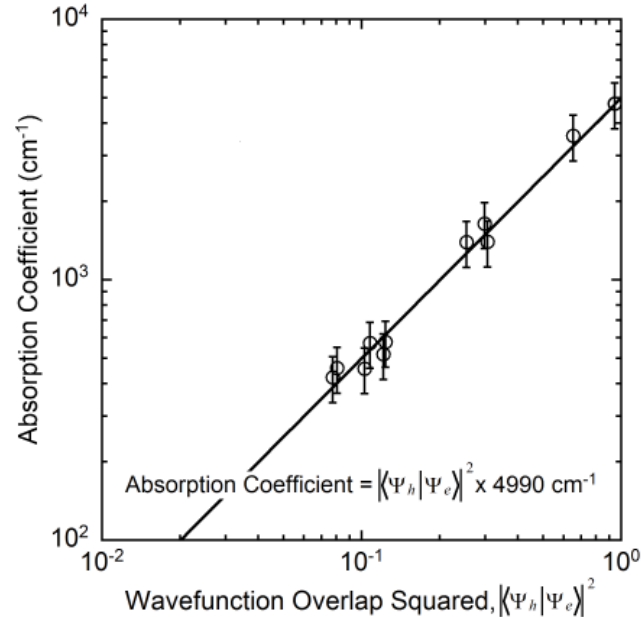
Thinner InGaAs layers act to improve coupling between adjacent hole wells, improving wavefunction overlap



Compare thicknesses ( $d$ ):  $d_{\text{InAs}} = 3 \times d_{\text{InAsSb}}$

$d_{\text{InGaAs}} = 0.9 \times d_{\text{InAsSb}}$

# Absorption Coefficient and Wavefunction Overlap



Transition strength (S) gives a measure of the strength of the optical transition independent of the optical joint density of states

$$S = \left( \frac{2}{m_0} \right) \cdot \frac{|\langle \Psi_h | \bar{\mathbf{p}} | \Psi_e \rangle|^2}{\hbar \omega} = \left( \frac{2m_0}{\hbar^2} \right) \cdot (\hbar \omega) \cdot |\langle \Psi_h | \bar{\mathbf{r}} | \Psi_e \rangle|^2$$

Transition strength can be determined from the measured absorption coefficient ( $\alpha$ ) by backing out the density of states ( $\rho$ )

$$S = \left( \frac{2m_0 n c \epsilon_0}{e^2 \pi \hbar} \right) \cdot \left( \frac{\alpha}{\rho} \right)$$

As wavefunction overlap approaches 100%, the absorption coefficient approaches  $4990 \text{ cm}^{-1}$  and the transition strength approaches 43.1

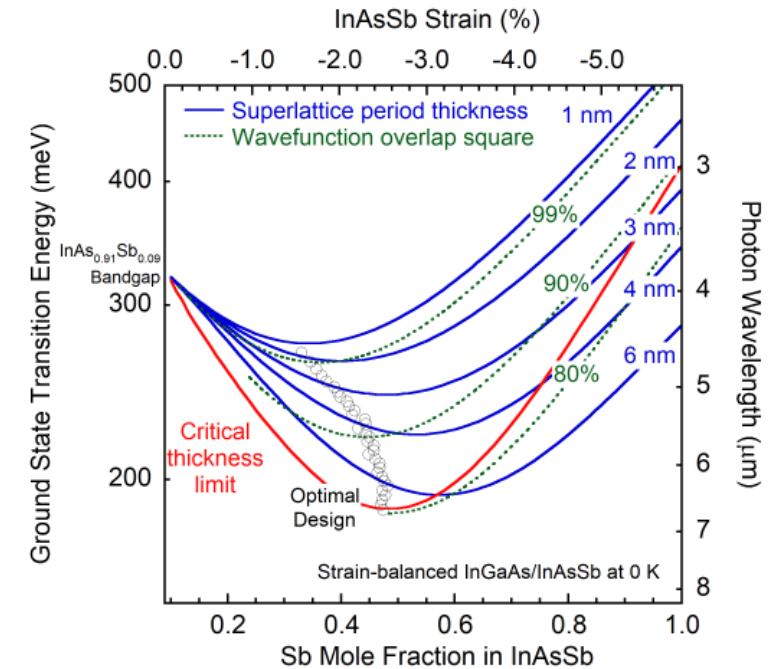
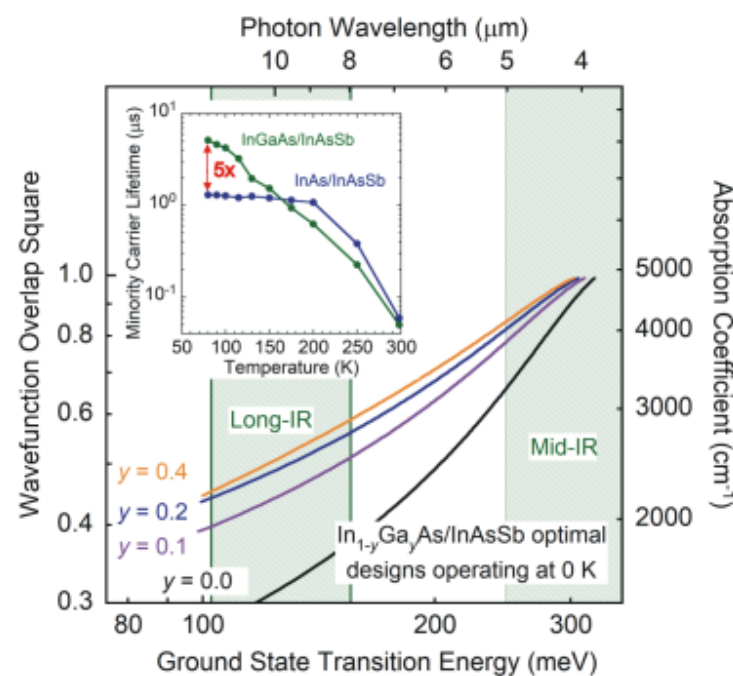
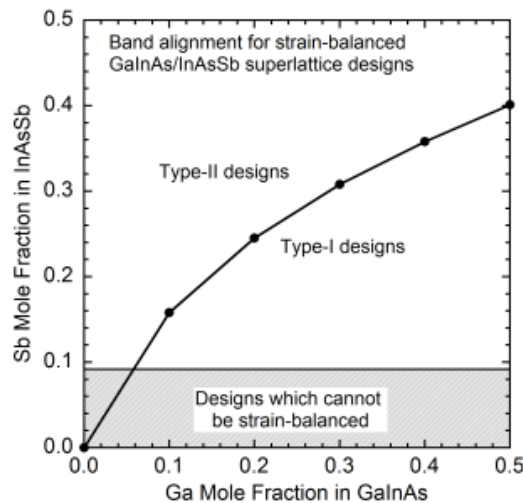
# General Observations and Optimal Design

More Sb is required to maintain type-II band alignment with increasing Ga mole fraction

The optimal design calls for short period thicknesses; the individual InGaAs and InAsSb layers are not expected to relax despite heavy strain

Some of the optimal designs are type-I

The optimal designs form ground state transitions through the heavy hole band



Increasing Ga in InGaAs improves wavefunction overlap via a more favorable strain-balance condition

~50% absorption increase in mid-IR, >2x increase in long-IR

# InGaAs/InAsSb Growth Conditions

InGaAs/InAsSb characterization structures are 1  $\mu\text{m}$  thick, doped to  $5 \times 10^{15} \text{ cm}^{-3}$  n-type

Narrower photoluminescence full width at half max observed at lower growth temperatures

Minority carrier lifetime  $\sim 2\text{-}3 \mu\text{s}$  at 100 K for all InGaAs/InAsSb samples

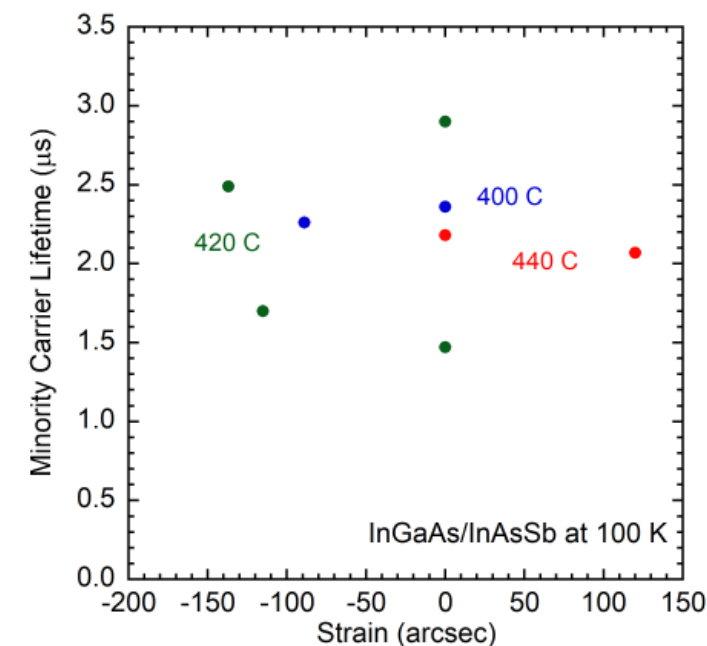
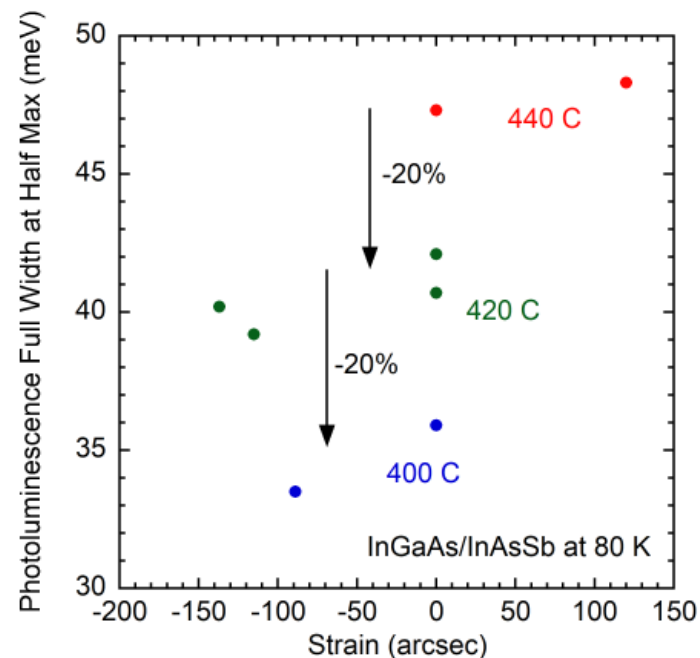
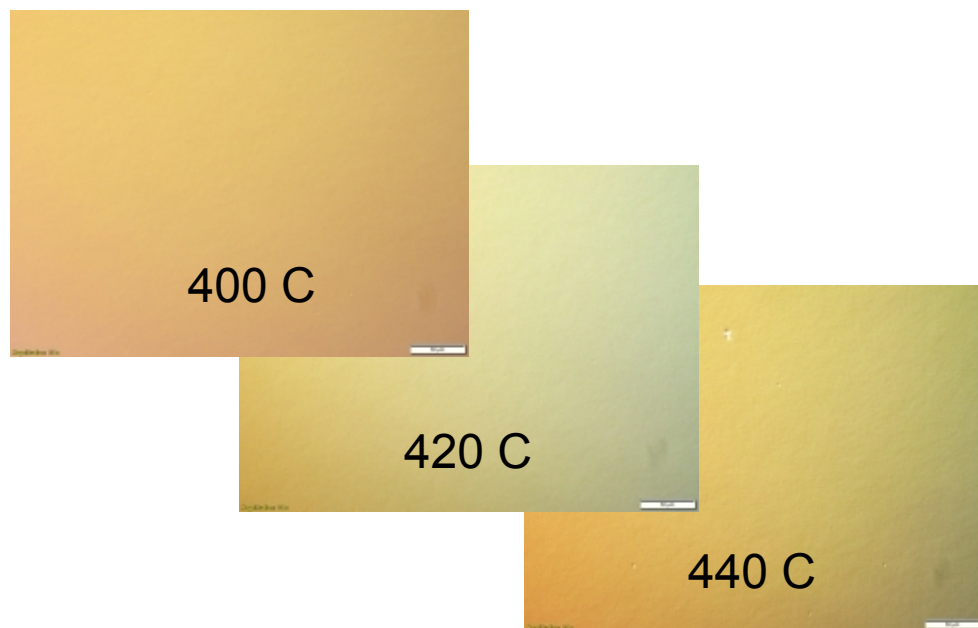
Minority carrier lifetime  $\sim 1 \mu\text{s}$  for  $5 \times 10^{15} \text{ cm}^{-3}$  n-type InAs/InAsSb grown under the same conditions

Confinement/Cap

1.0  $\mu\text{m}$  n-type  
InGaAs/InAsSb  
superlattice

Buffer/Confinement

n-type GaSb  
substrate

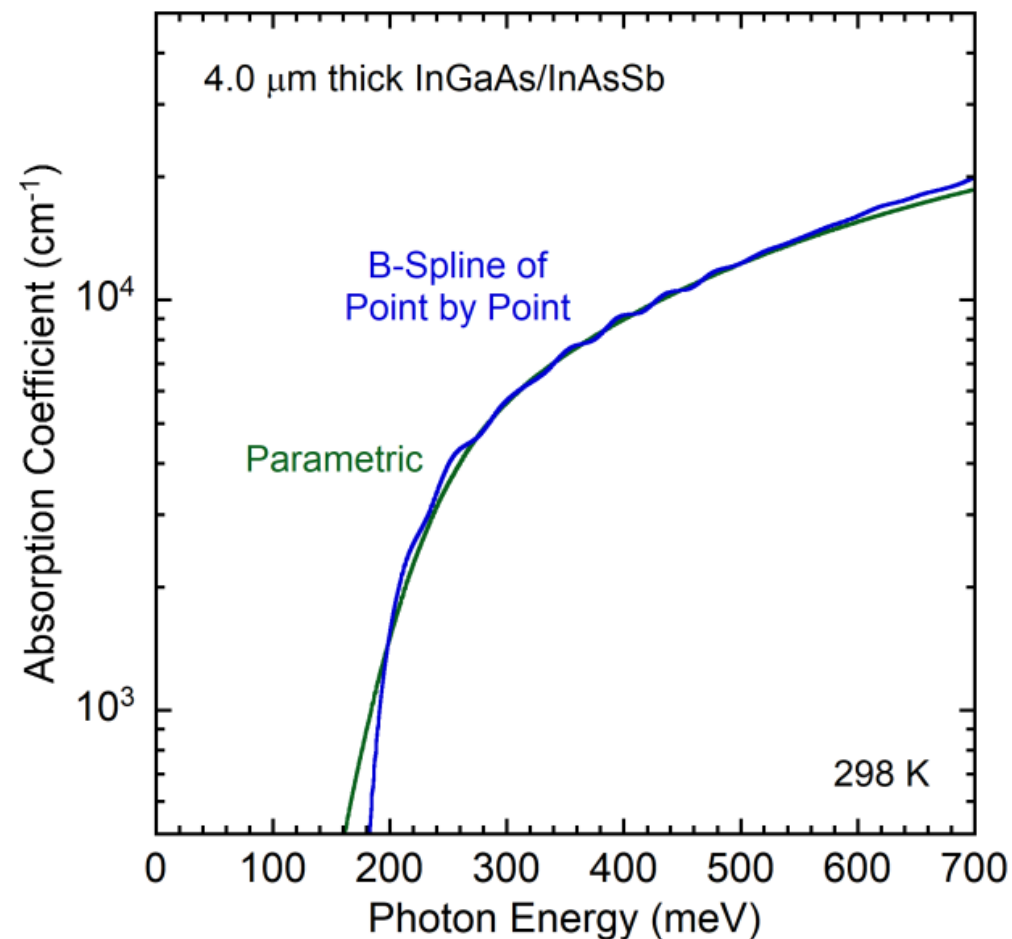
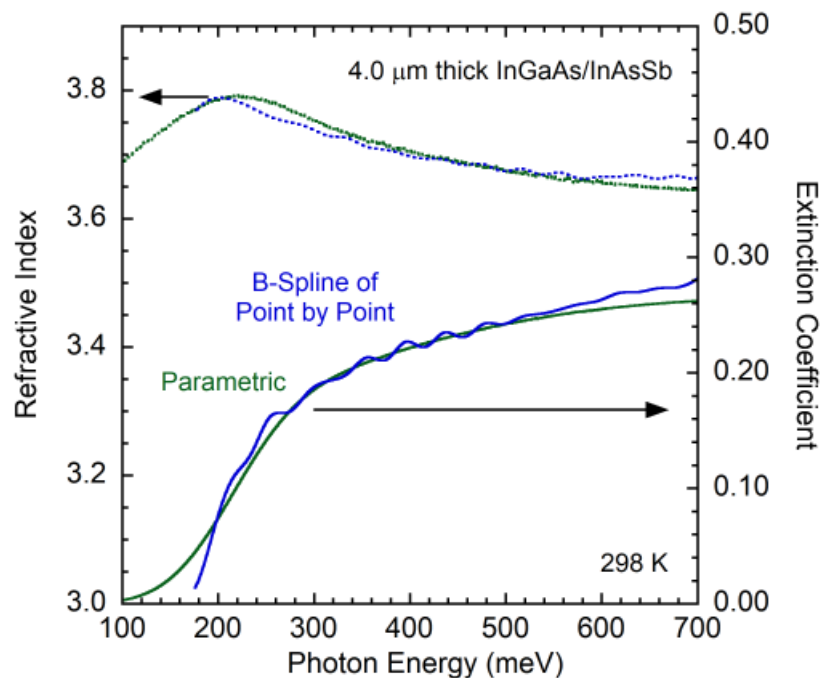




# InGaAs/InAsSb Optical Constants

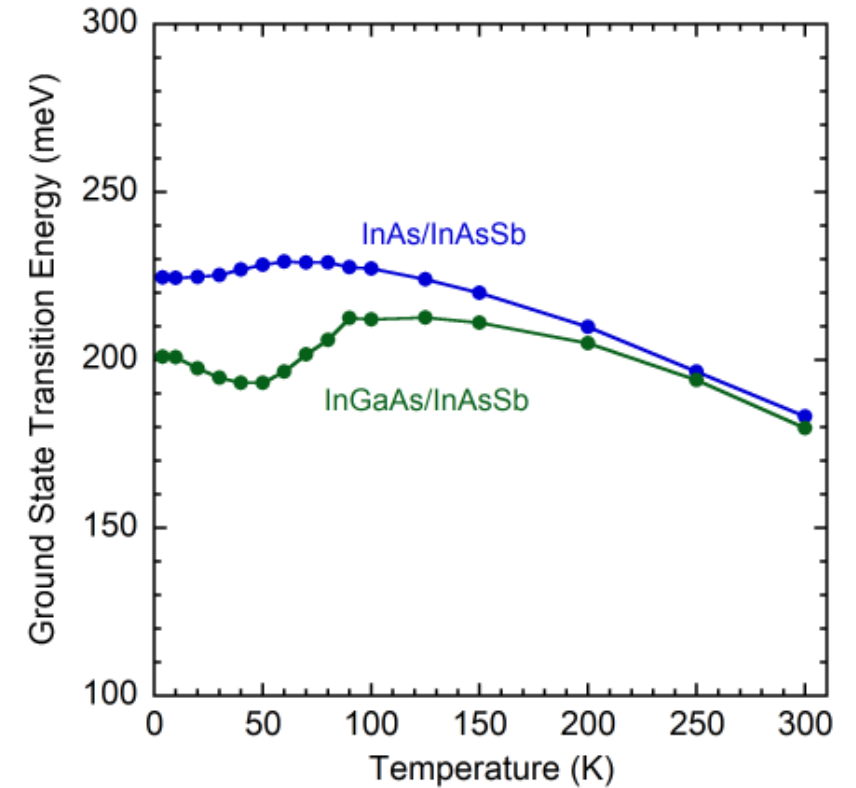
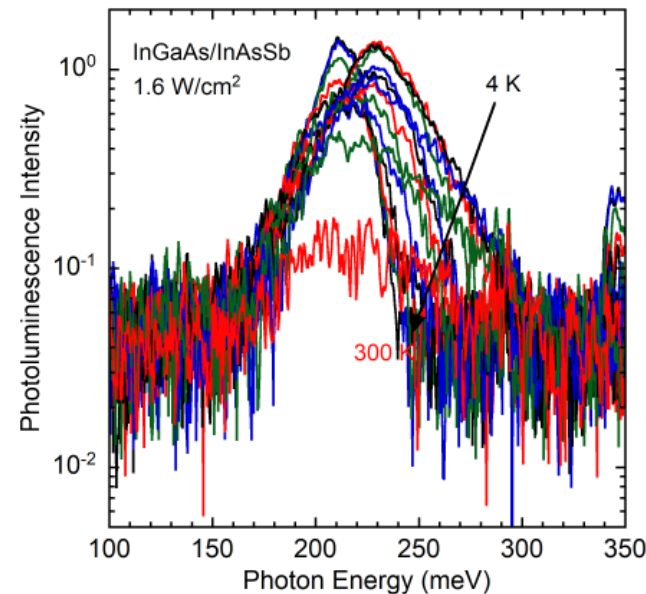
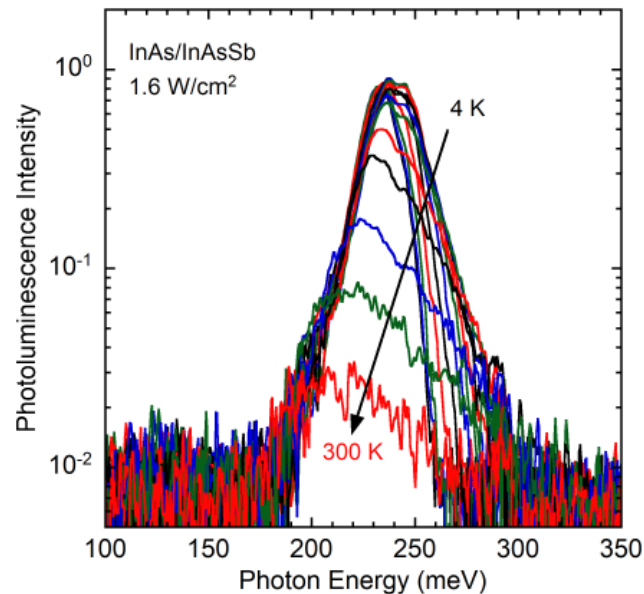
Optical constants of 4.0  $\mu\text{m}$  thick layer of mid-wave InGaAs/InAsSb measured by spectroscopic ellipsometry

Ground state transition energy shifts to 177 meV (7  $\mu\text{m}$  wavelength) at room temperature



# Steady-State Photoluminescence

Photoluminescence shifts to lower energies below 100 K in the InGaAs/InAsSb superlattice, possibly indicating the presence of localized states



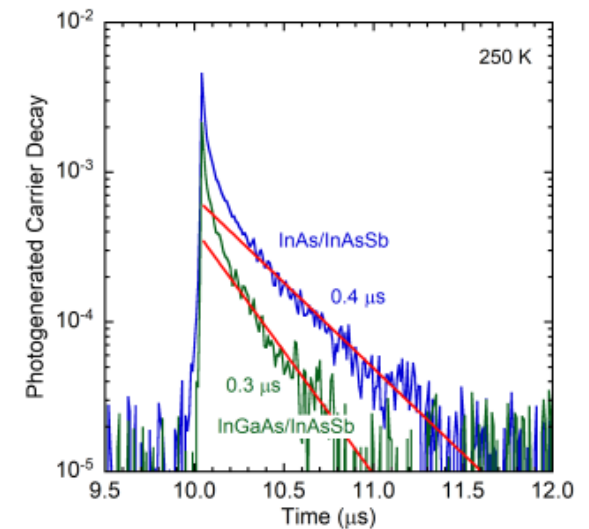
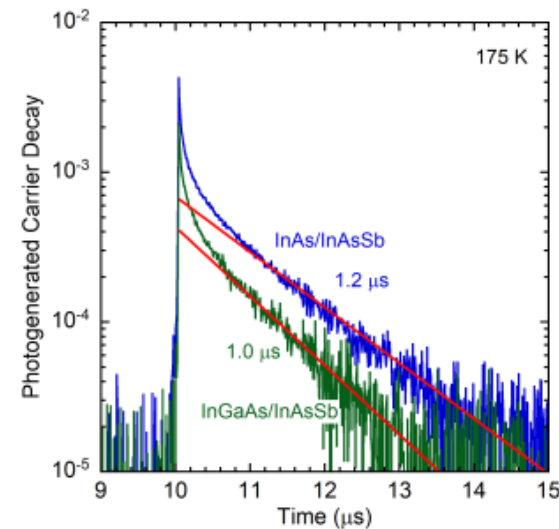
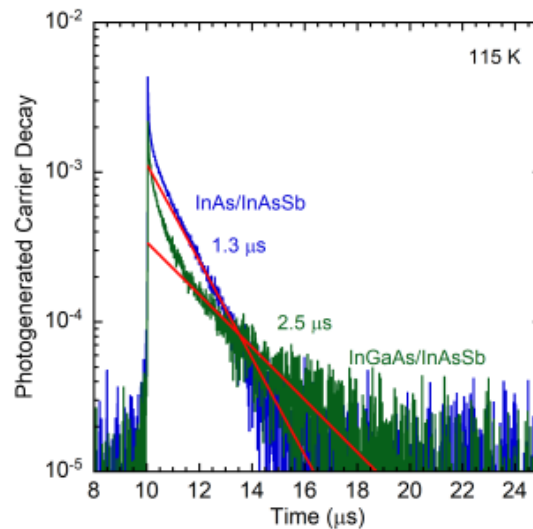
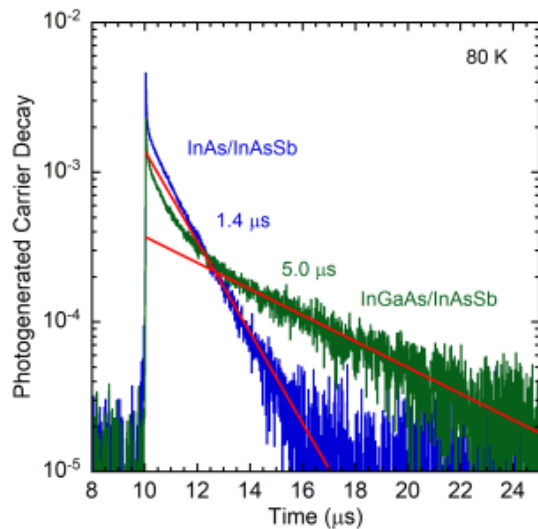
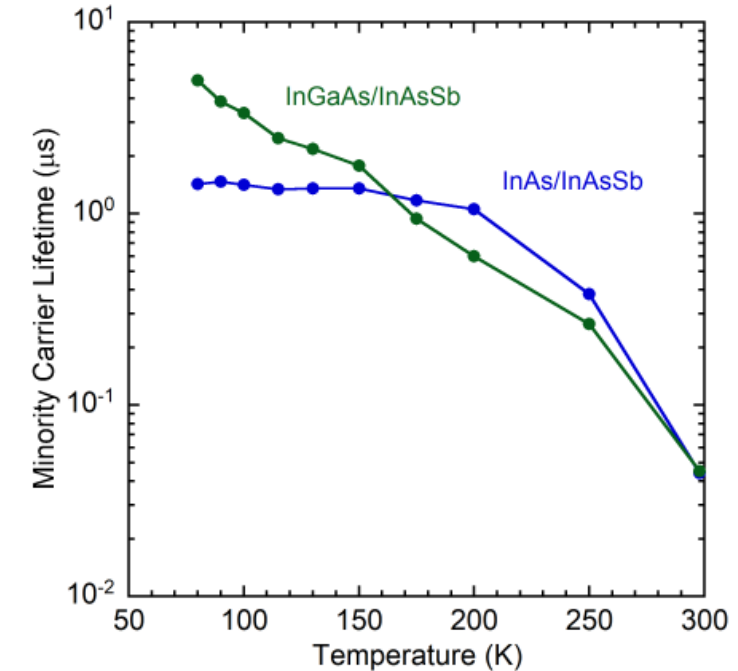


# Transient Microwave Reflectance

A microwave probe beam is incident on the sample; the reflected microwave power is a function of the sample conductivity

Samples are excited by a short pulse 1535 nm laser, and the photogenerated carrier population modifies the sample conductivity

The conductivity and corresponding microwave reflectance decay with the photogenerated carrier population

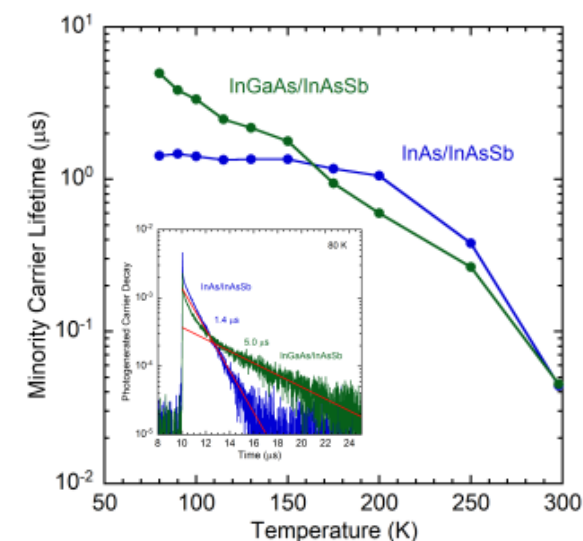
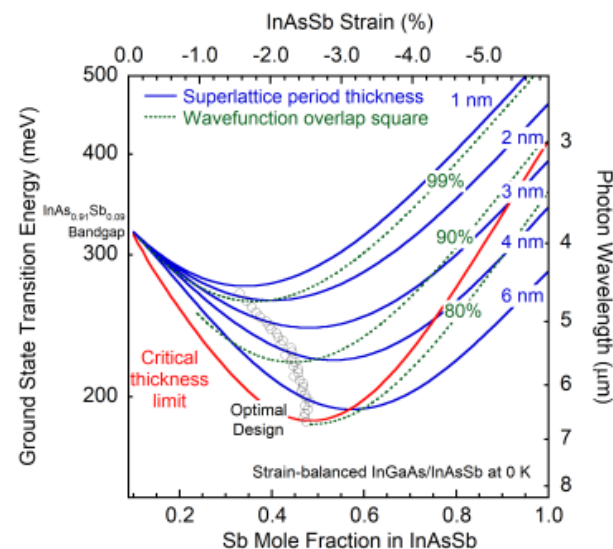
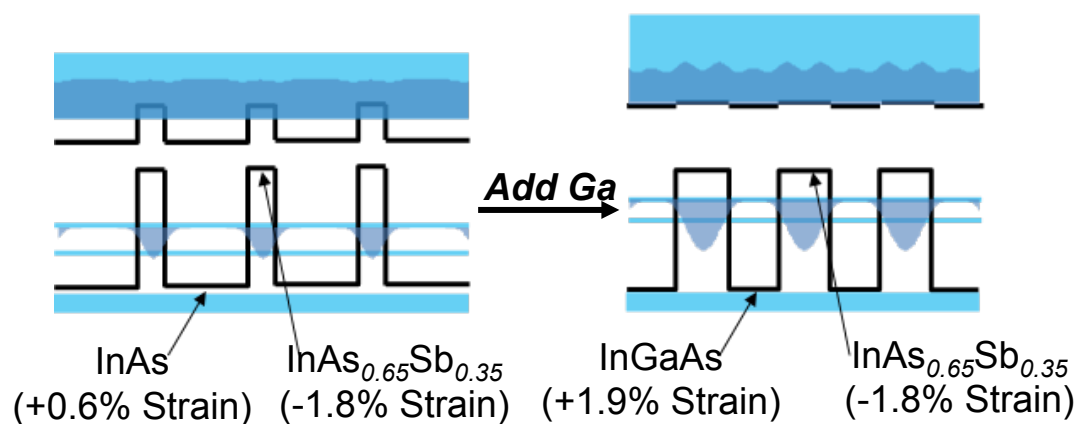


# Conclusions

Adding Ga to InGaAs/InAsSb superlattices yields more tensile InGaAs, resulting in a more symmetric strain-balance profile and better wavefunction overlap and absorption

Molecular beam epitaxy growth conditions identified for smooth, high-quality mid-wave InGaAs/InAsSb with long minority carrier lifetime and narrow photoluminescence full-width at half max

Si-doped InGaAs/InAsSb exhibits a 5× longer minority carrier lifetime than identically doped InAs/InAsSb



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