

# Understanding Governing Factors of Localized Corrosion in Marine Environments

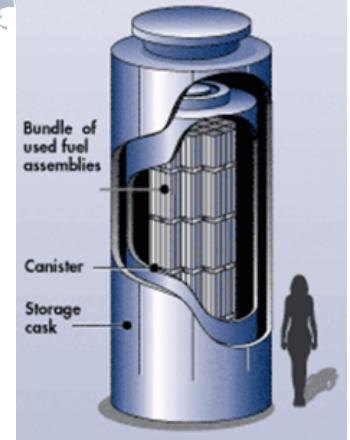
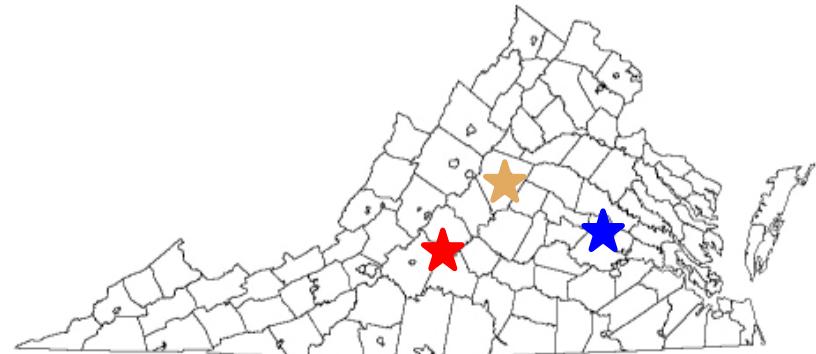
R. M. Katona<sup>1,2</sup>

<sup>1</sup>Materials Science and Engineering, University of Virginia, Charlottesville, Virginia 22904

<sup>2</sup>Sandia National Laboratories, Albuquerque, New Mexico 87123, USA

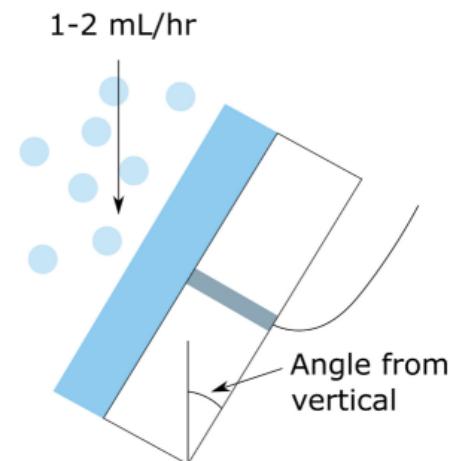
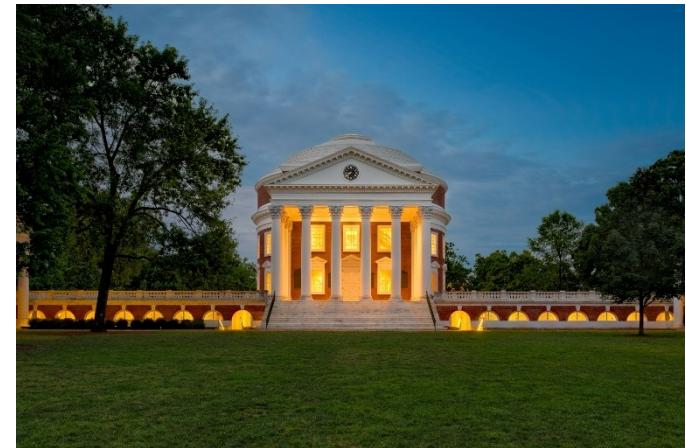
# About me

- From right outside Richmond, VA
- Bachelor of Science in Physics and Mathematics – 2017
  - University of Lynchburg (VA)
- Master of Engineering in Materials Science and Engineering – 2019
  - University of Virginia (UVA)
- PhD in Materials Science and Engineering – Expected Summer 2021
  - Anodic and Cathodic Limitations on Localized Corrosion and Stress Corrosion Cracking Propagation of Stainless Steel 304L in Atmospheric Environments
- Graduate Student Intern
  - Storage and Transportation technology



# About Graduate School

- Advisor: Dr. Robert Kelly
- Center for Electrochemical Science and Engineering (CESE)
- Outreach
  - Creation of 3D printing class for 4-5<sup>TH</sup> graders at the UVA Curry School for Education
- Consulting at UVA
  - Assisted in creating a Virtual Twin for a spent nuclear fuel canister
  - Incorporation of pit modeling into twin to identify potentials for materials degradation
- Highlighted Research (not presented today)
  - Creation of sensor to determine water layer thickness in accelerated testing
- Publication record
  - 9 publications accepted/submitted (7 first author)

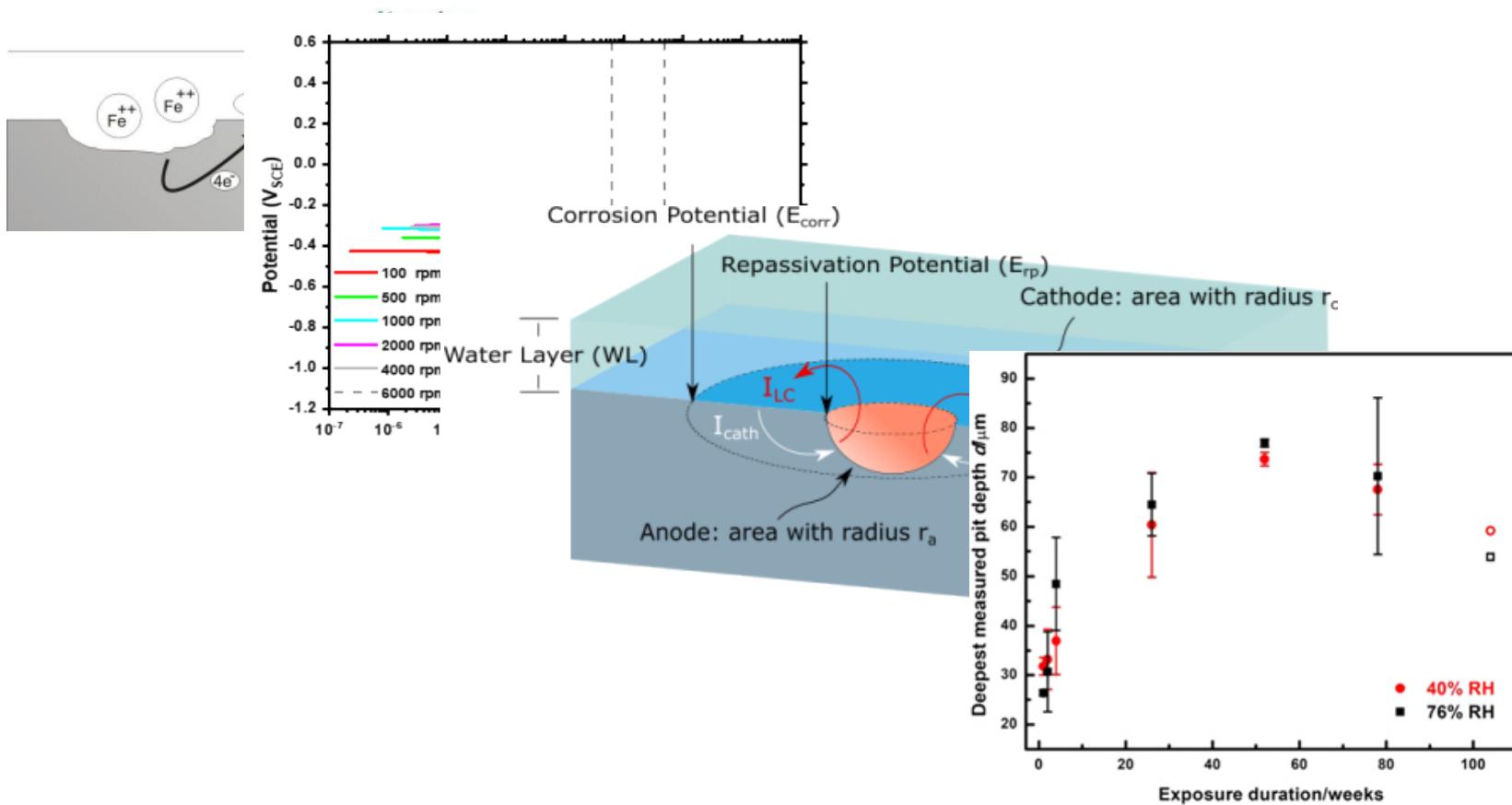
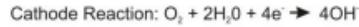


# At Sandia

- Advisor: Dr. Rebecca Schaller (Materials Reliability)
  - Previously Dr. Eric Schindelholz
- Working in Storage and Transportation Technology
- Proposals
  - Three proposals written with one successfully funded
- Creation of laboratory space
  - Lead environmental assessments, failure modes analysis, and created working documents
- Experimental design and testing on electromechanical and hydraulic fracture mechanics load frames
- Advised multiple undergraduate students from University of New Mexico
  - Taught fundamentals of electrochemistry
  - Helped create posters for presentation at conferences

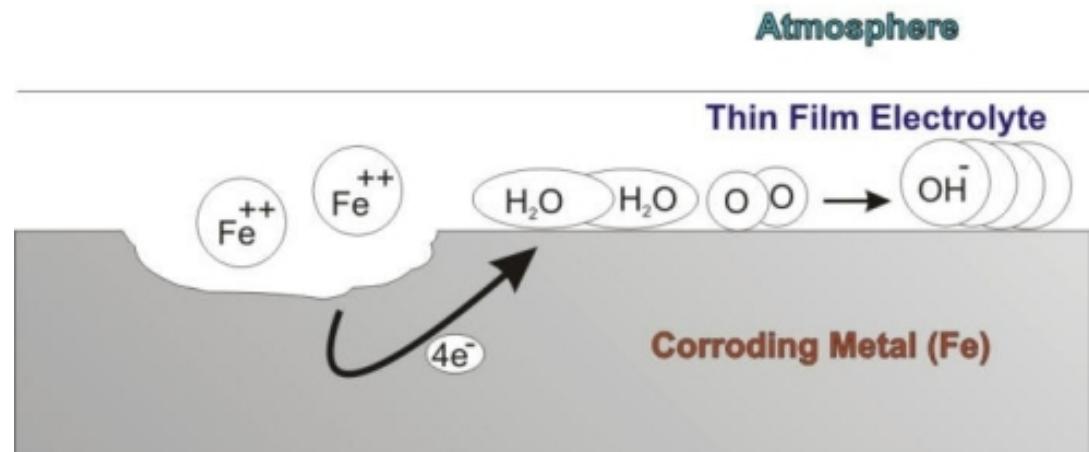
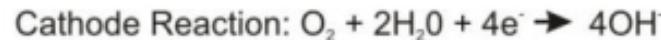
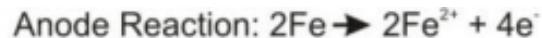


# Overview



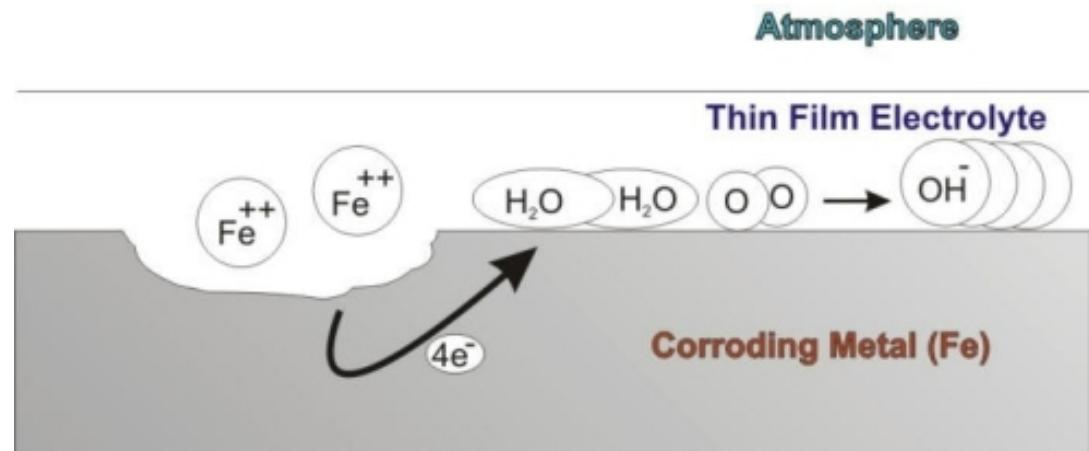
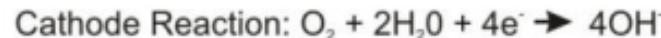
# Atmospheric corrosion

- Alloys are commonly exposed to atmospheric, marine environments
- Through salt enabled deliquescence or salt spray the creation of a water layer (WL) on the surface of an alloy allows for a corrosion cell to form
- Many factors will influence the corrosion
  - WL
  - Solution composition
  - Solution concentration
  - Temperature
  - Alloy
  - Reaction mechanisms



# Atmospheric corrosion

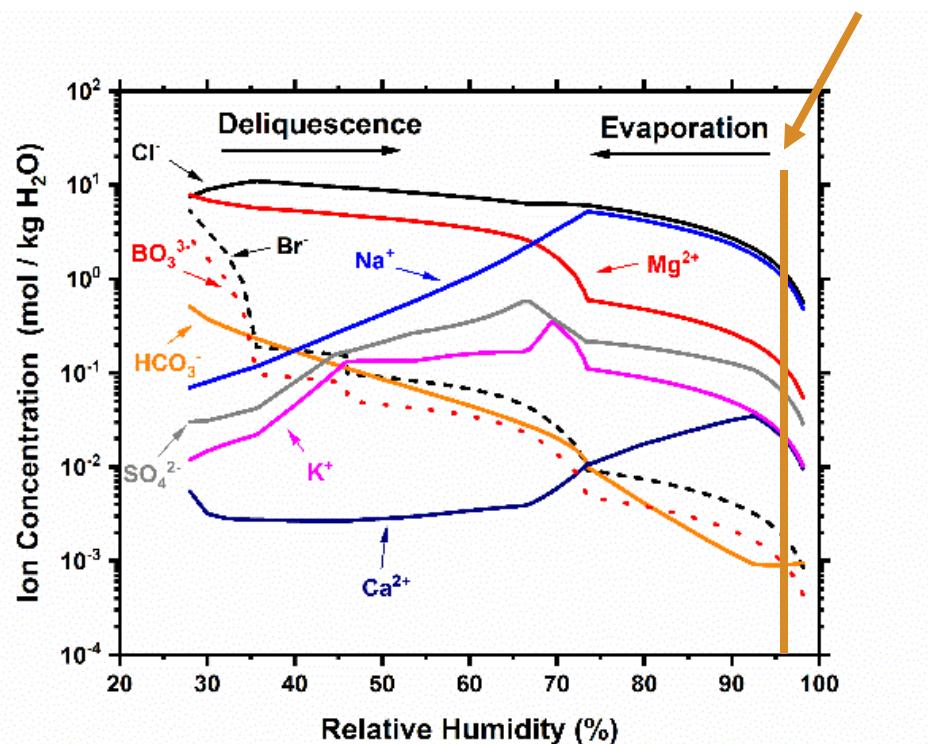
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# Atmospheric Brines

- Dehydration of seawater brine shows wide variation in composition
- Important relative humidities:
  - Precipitation of NaCl  $\sim$  75 % RH
  - Precipitation of MgCl<sub>2</sub>  $\sim$  35 % RH

98 % RH  $\sim$  0.6 M NaCl

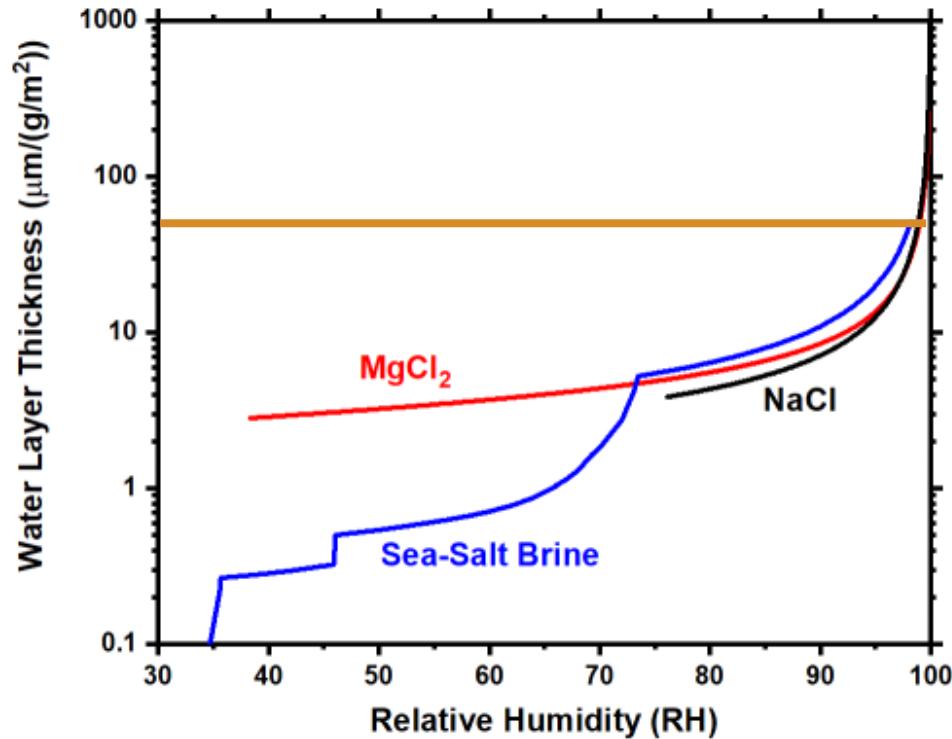


# Thin Atmospheric Water Layers

- In atmospheric scenarios, salt enabled deliquescence is possible
- WL dependent upon:
  - Salt composition
  - Relative humidity
  - Temperature
  - Loading density

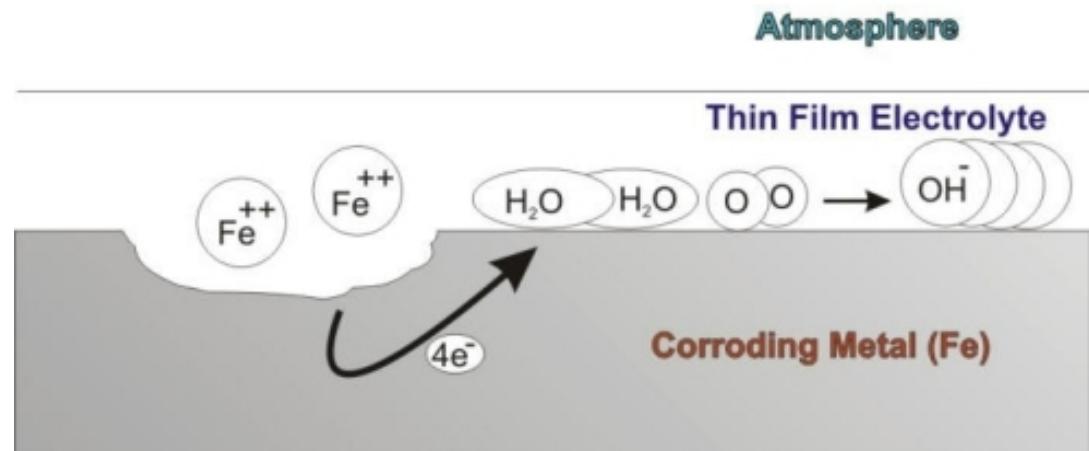
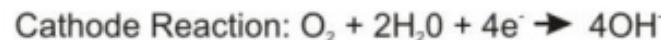
$$WL = \frac{LD * \rho_{sp}}{MW * C_{eq}}$$

- Under typical conditions a  $WL < 200 \mu\text{m}$  is expected
- Due to condensation, rain, or salt spray, WLs above  $1,000 \mu\text{m}$  are likely, at least transiently



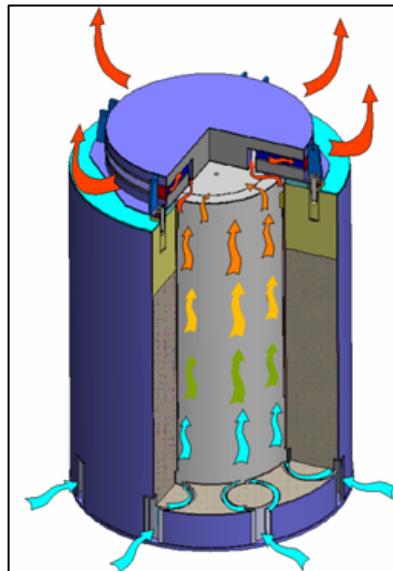
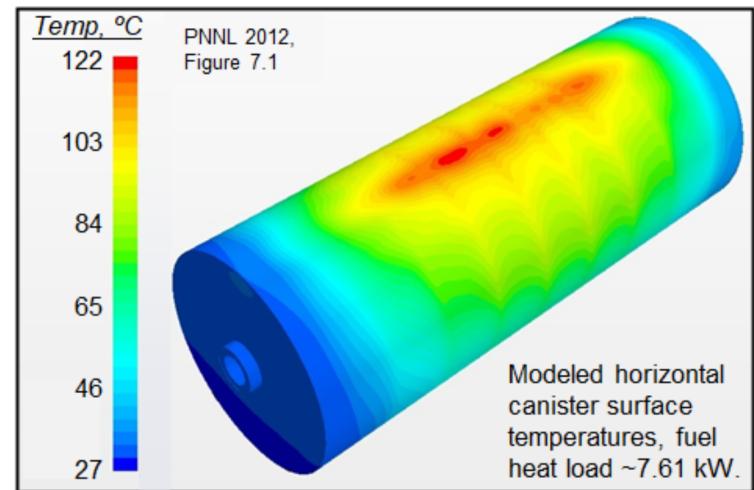
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  - Alloy
  - Reaction mechanisms

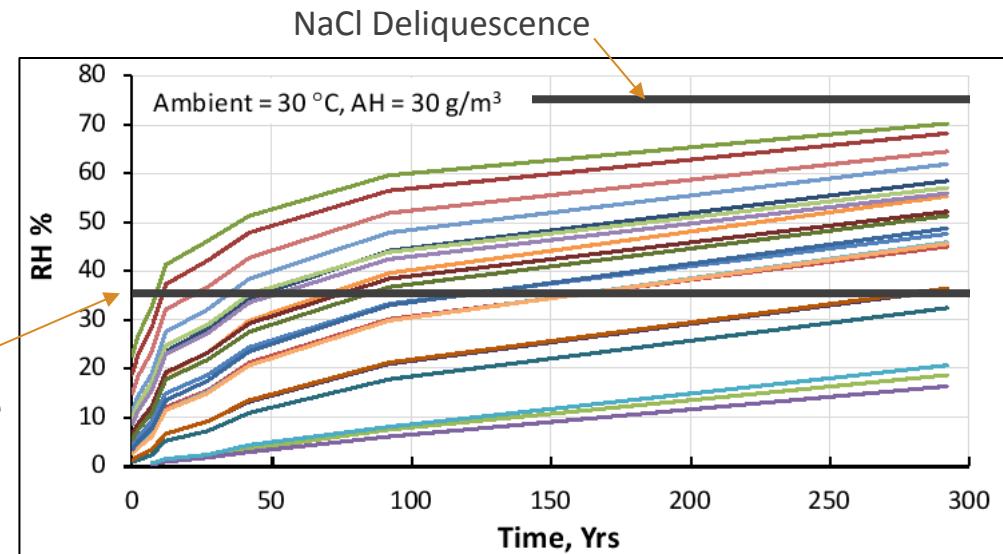


# Temperatures

- Seasonal and diurnal fluctuations of temperature
- One specific application of work is for spent nuclear fuel storage
- Combination of ambient temperatures, ambient RH, and surface temperature

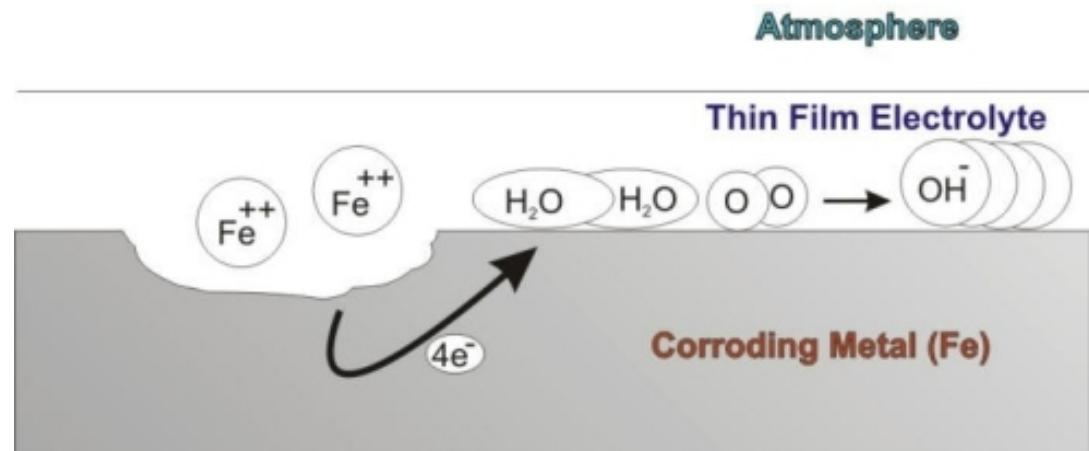
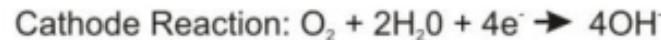
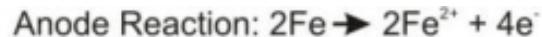


FCRD-UFD-2012-000114 Figure 7.3



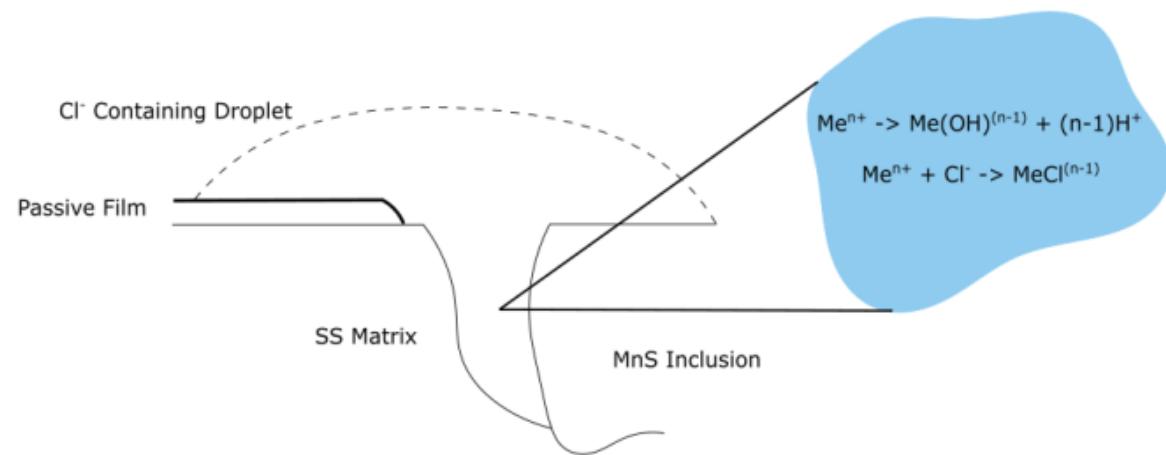
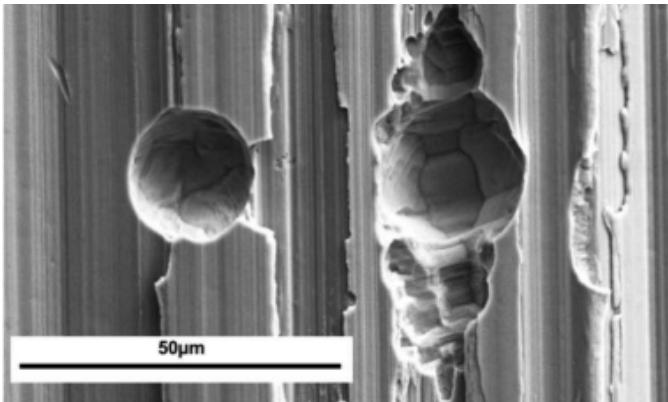
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  - *Reaction mechanisms*



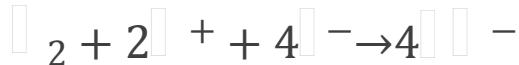
# SS Susceptibility to Localized Corrosion

- Failure mechanisms for SS
  - Pitting corrosion, crevice corrosion, and stress corrosion cracking
- Pitting corrosion typically initiated at magnesium sulfide (MnS) inclusions
- Creation of an aggressive environment through metal ion hydrolysis

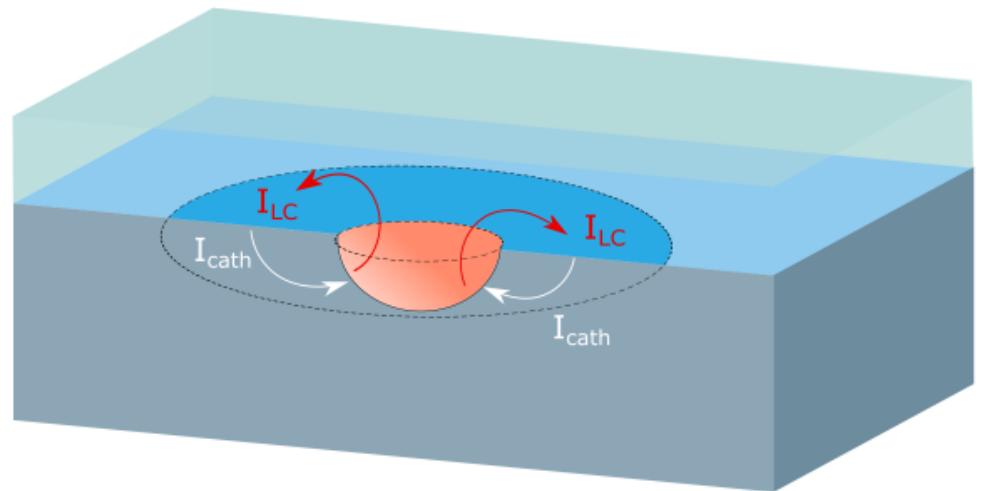


# SS Reaction Mechanisms

- There are various mechanisms for pit initiation, however always will have an aggressive environment in the pit
- Dissolution supported by cathodic reduction reaction
- In atmospheric scenarios typically thought to be oxygen reduction reaction

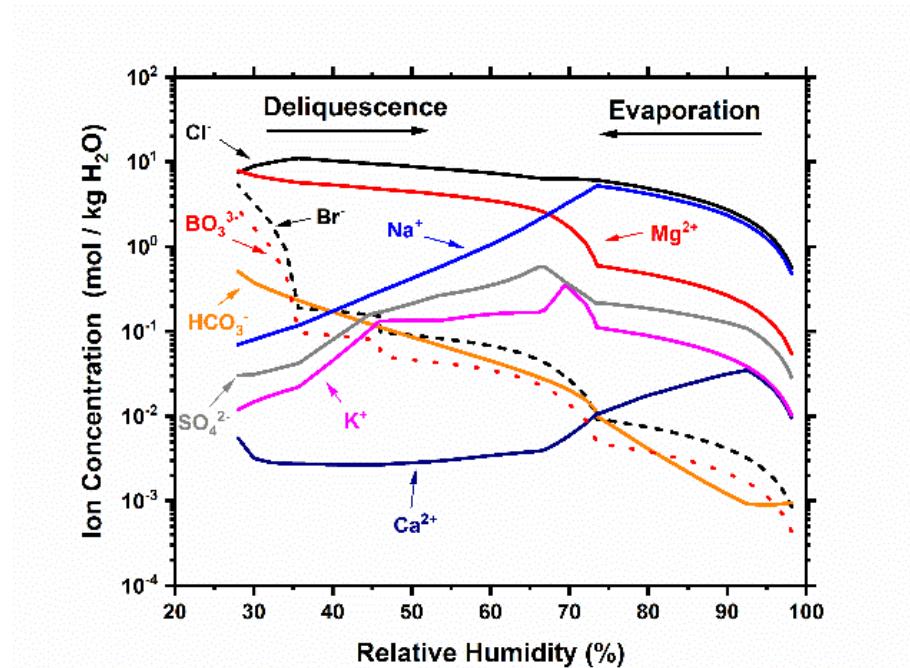


- Literature determined reaction mechanisms focus on dilute solutions typically at room temperature



# What is missing in localized corrosion?

- Corrosion reaction mechanisms in high chloride containing solutions across a wide range of relative humidity and temperature



- Prediction and validation of localized corrosion in atmospheric environments containing various solution properties

# Critical Dissertation Questions

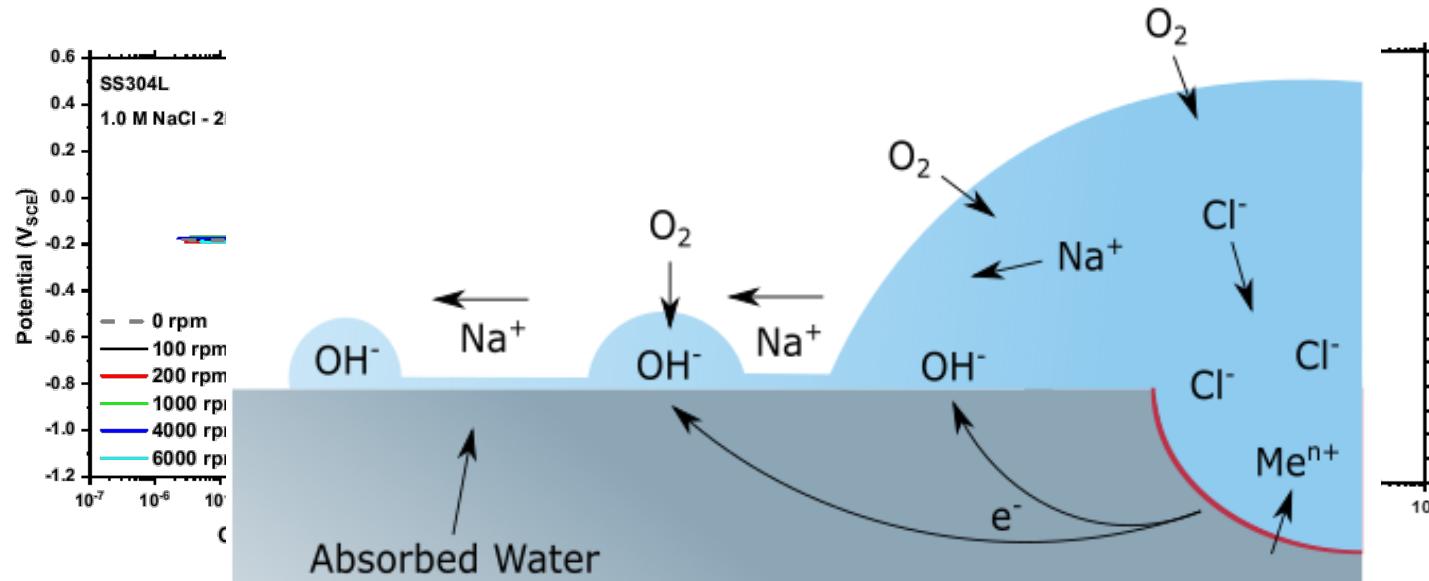
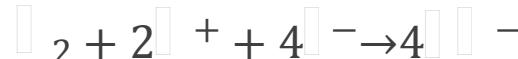
- What mechanisms control how external (solution composition, solution concentration, geometry, temperature) and internal (material) factors affect pitting and stress corrosion?
- To what extent can we predict damage from localized corrosion and stress corrosion cracking and what are the governing factors in these predictions?

# Critical Dissertation Questions

- What mechanisms control how external (solution composition, solution concentration, geometry, temperature) and internal (material) factors affect **pitting and stress corrosion**?
- To what extent can we predict damage from localized corrosion and stress corrosion cracking and what are the governing factors in these predictions?

# Cathodic Reaction Mechanisms

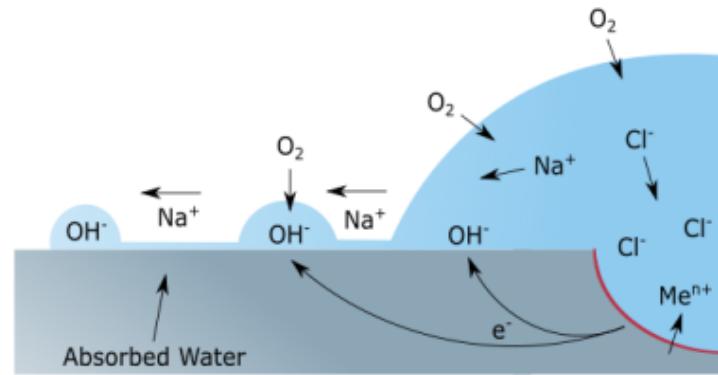
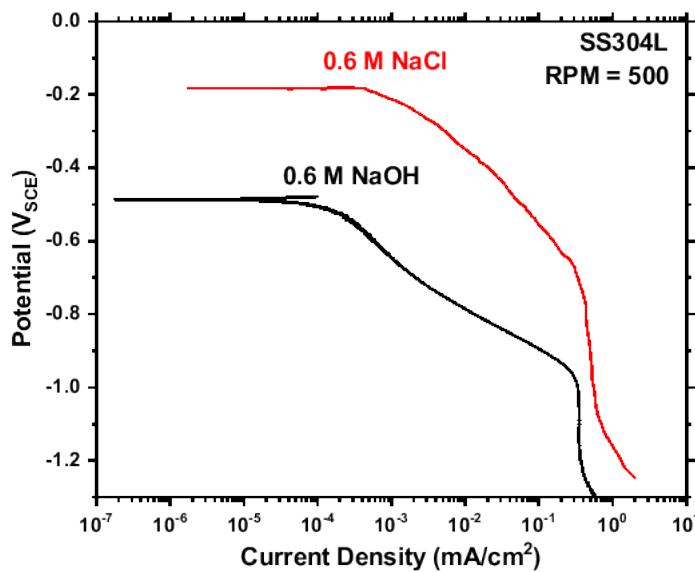
- For NaCl solutions (high RH), oxygen reduction (ORR) is dominant cathodic reduction reaction



- Holds for all concentrations (0.6 – 5.3 M) and temperatures (25–45 °C) of NaCl explored

# Cathodic Reaction Mechanisms

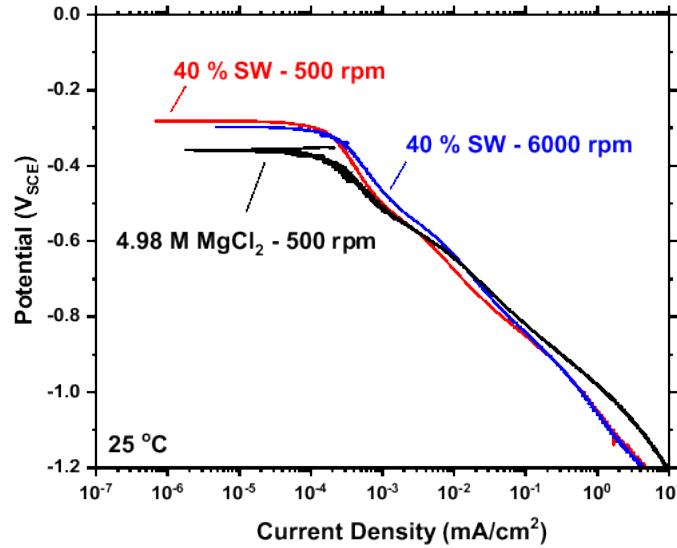
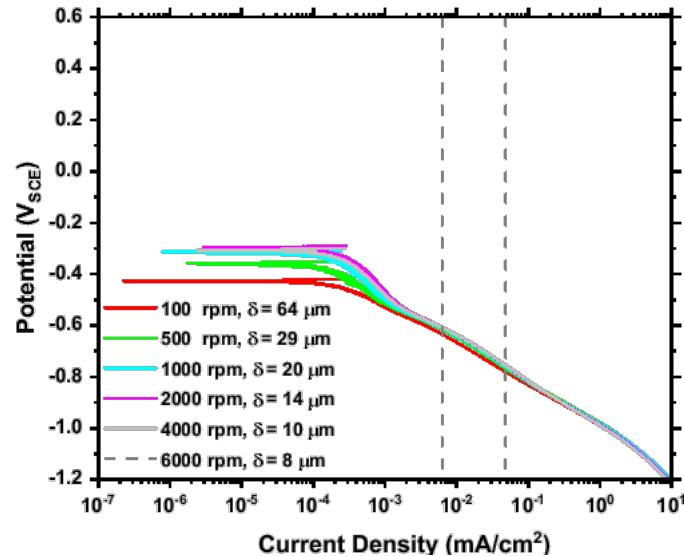
- NaOH solutions (evolved cathode of NaCl) exhibit one-electron transfer ORR as rate limiting step



- ORR suppression in NaOH solutions
- Decrease in current density in NaOH solution in comparison to NaCl solutions

# Cathodic Reaction Mechanisms

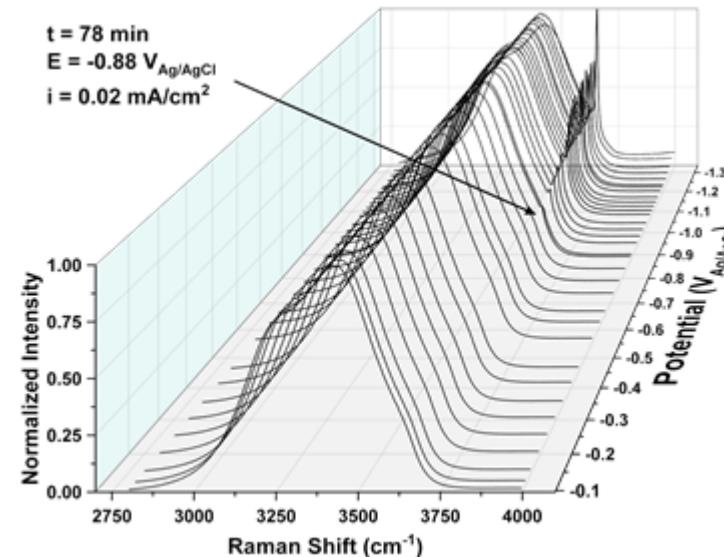
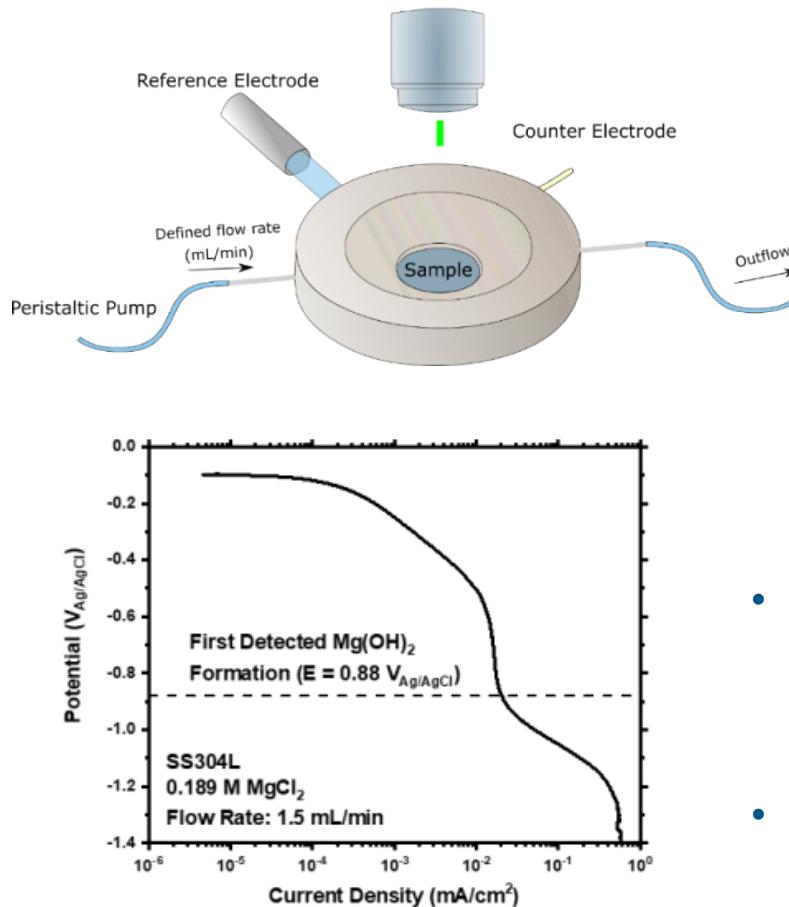
- Hydrogen evolution (HER) dominant in  $\text{MgCl}_2$  brines (low RH)
- Also present at elevated temperatures and for seawater solutions



- Proposed HER due to ORR suppression due to precipitate formation in brine and localized corrosion

# In-situ Spectroelectrochemistry

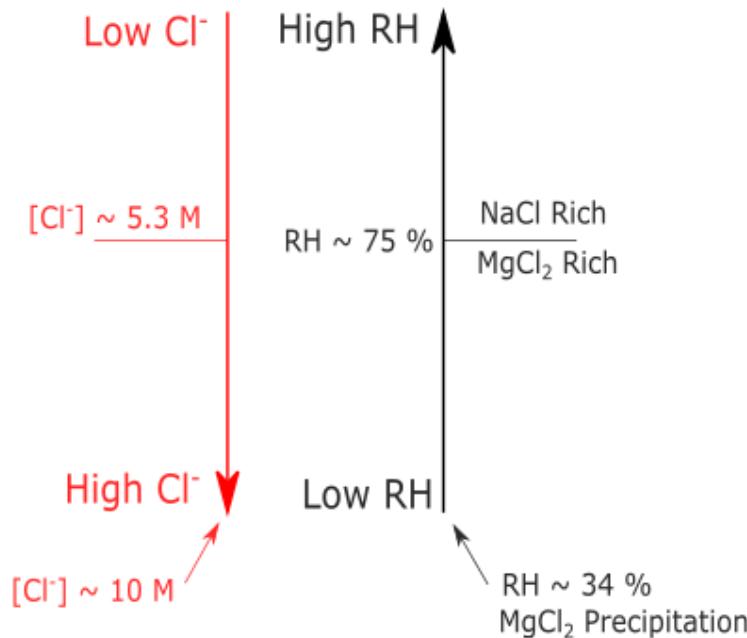
- Visualization of precipitate formation in  $MgCl_2$  solutions as a function of cathodic polarization



- $Mg(OH)_2$  identified
  - ORR suppression
  - $MgCO_3$  not kinetically stable
- Technique development to control boundary layer with flow rate

# Cathodic Kinetics Summary

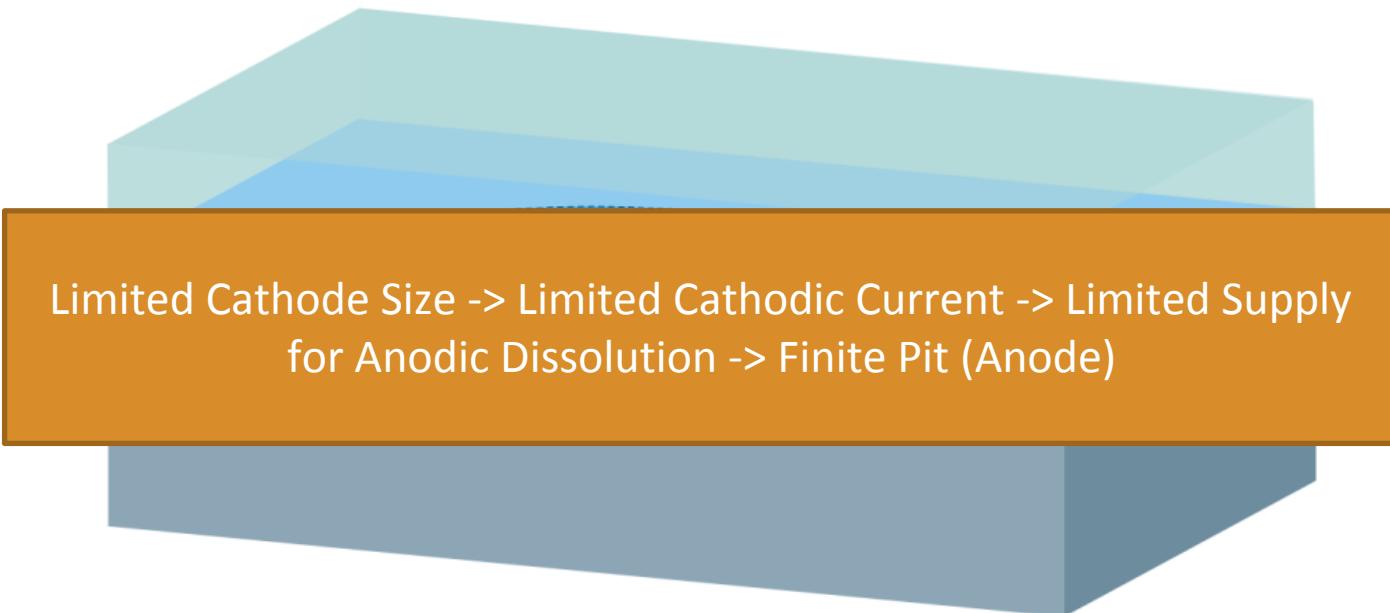
- Across wide range of solutions, various mechanisms are dominant



- Now that we know mechanisms, we can model corrosion scenarios

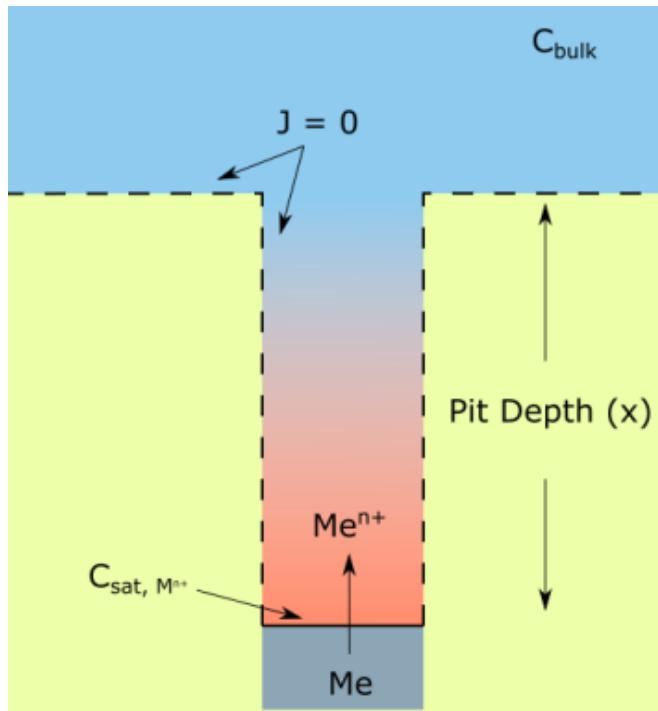
# Modeling Localized Corrosion

- In chloride environments, pitting is highly probable, therefore worried about **maximum extent of corrosion**
- Pits (anode) are inherently coupled to the surrounding cathode

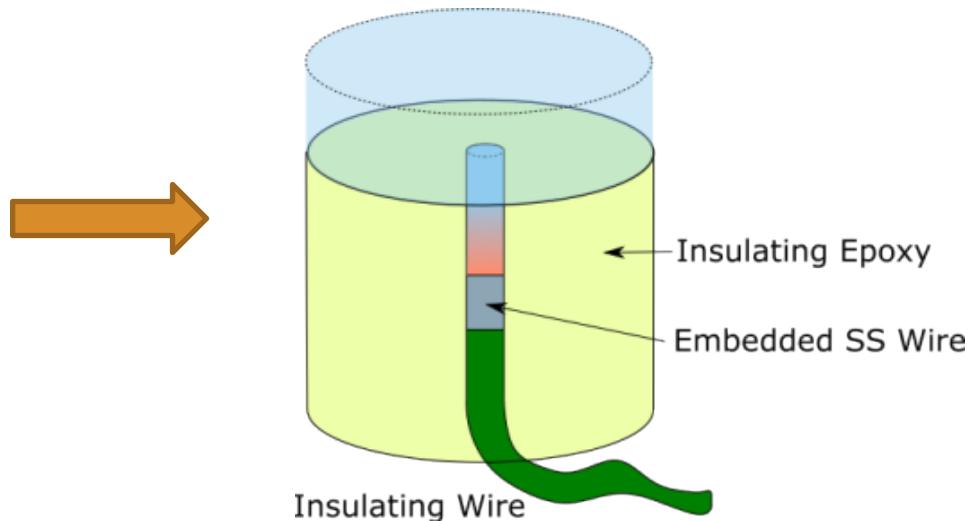


- Anode can only grow if sufficient supply from cathode
- Cathode current limited by reaction mechanism and physical geometry (water layer, sample size, etc.)

# Background – Pit Propagation



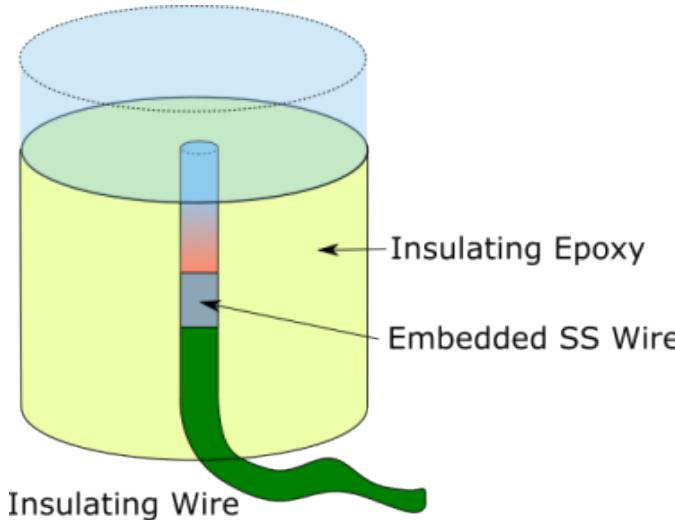
Studied through lead-in-pencil experiments under a salt film (full saturation) [1]



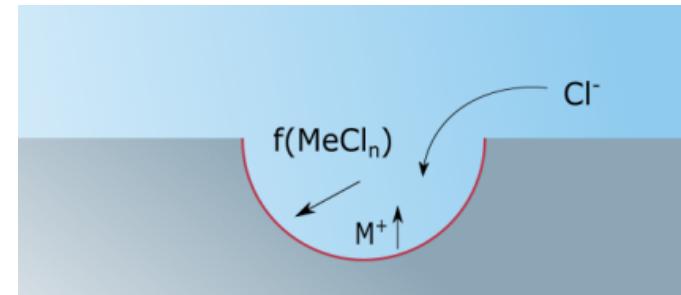
Schematic of 1-D scenario with active alloy surrounded by no flux ( $J$ ) boundary ( $J=0$ )

- Battle between outward ion diffusion and maintaining critical environment
- Galvele 1-Dimensional analysis yielded a pit stability product ( $i \cdot x$ ) [2]
  - Where  $i$  is current density and  $x$  is the pit depth

# Background – Pit Propagation



Converted to hemisphere by a geometric factor of 3



- To sustain pitting, the current ( $I$ ) at a given radius ( $r$ ) must satisfy

$$\left(\frac{I}{r}\right) \geq \left(\frac{I}{r}\right)_{crit}$$

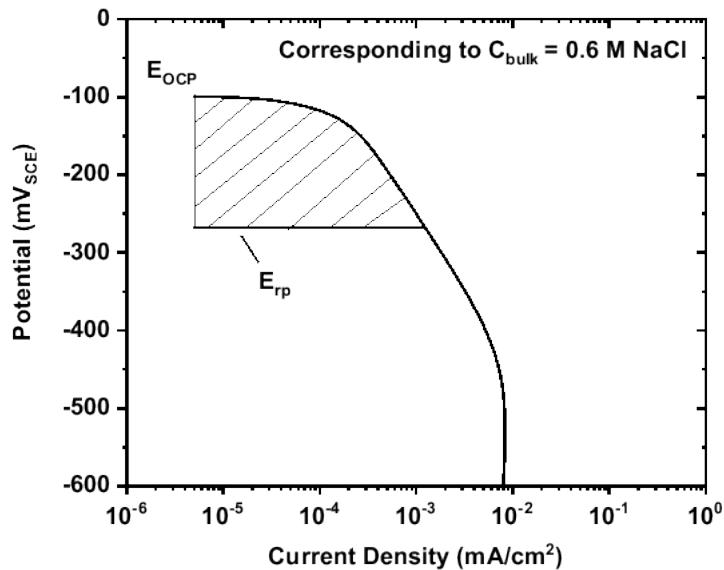
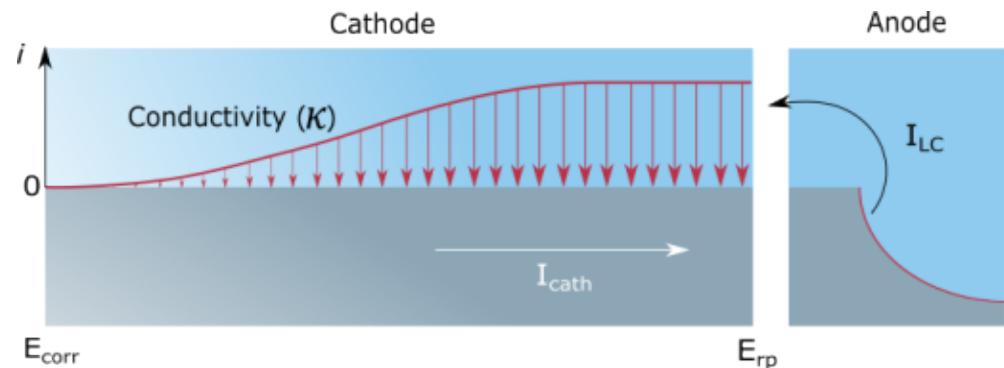
- Critical value  $\sim 50\%$  of full saturation of salt film needed

- Limiting anodic current demand ( $I_{LC}$ ):

$$I_{LC} = \left(\frac{I}{r}\right)_{crit} r_{anode}$$

# Cathode Supply

- Ohmic drop in cathode governed by current densities, water layer thickness, and conductivity

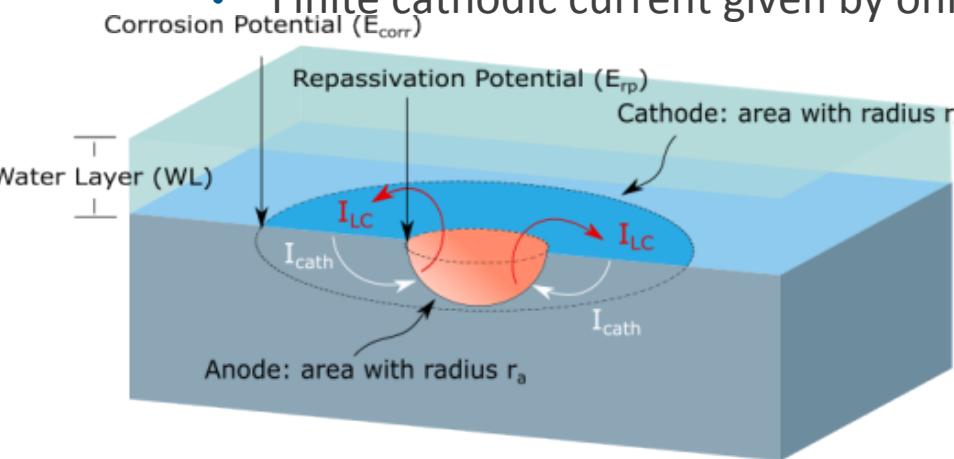


$$I_{c,max} \sim f(\kappa, WL, E_{OCP}, E_{RP}, r_a, i_{eq})$$

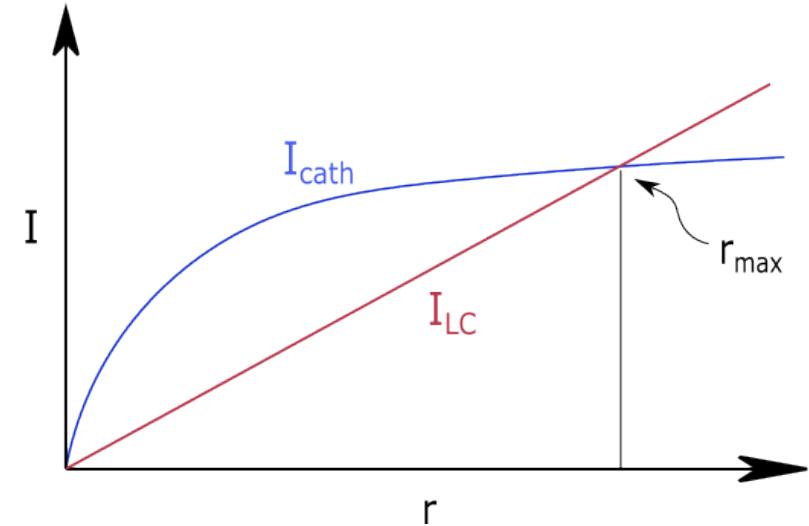
- Cathode current function of conductivity, WL, electrochemical potentials, anode radius, and equivalent current density

# Modeling Localized Corrosion

- Putting it all together
  - Anodic demand determined by ability to maintain aggressive environment
  - Finite cathodic current given by ohmic drop in thin electrolyte layers

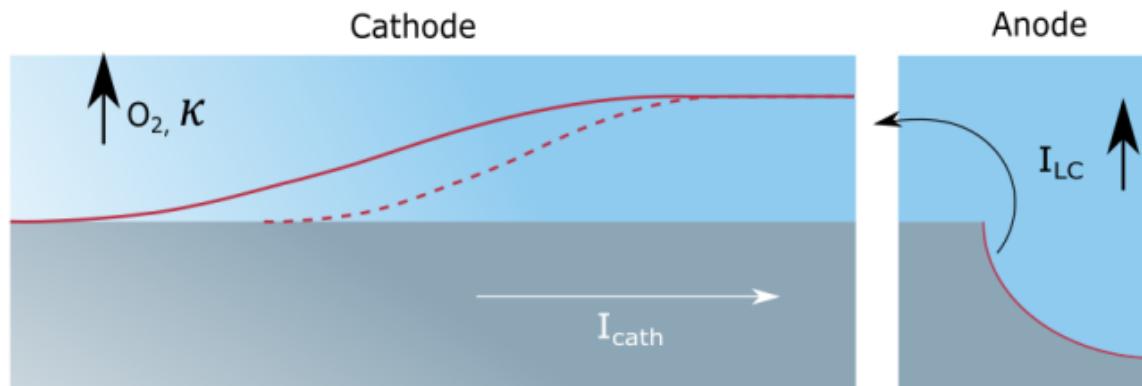
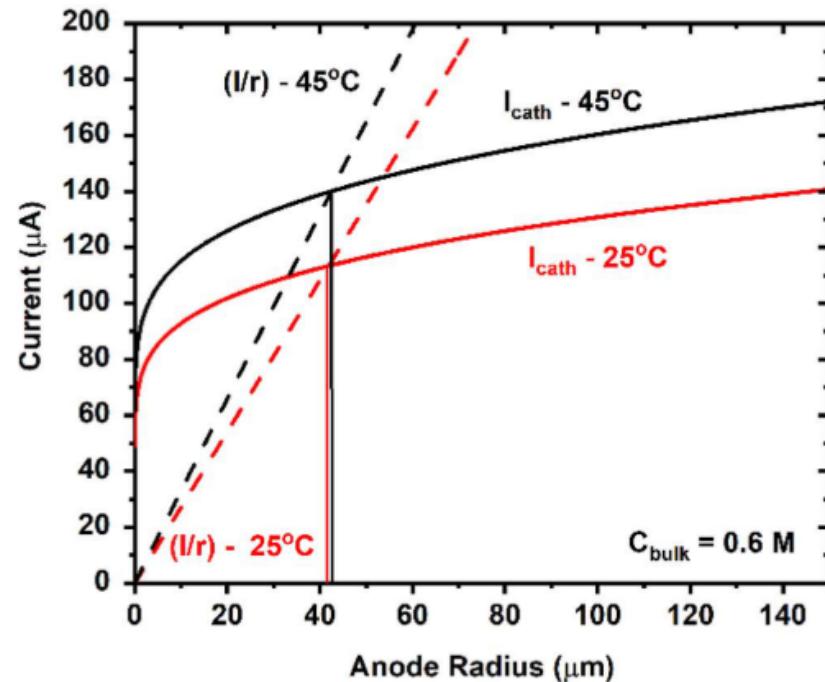


Anodic Demand = Cathodic Supply  $\rightarrow$  Max pit



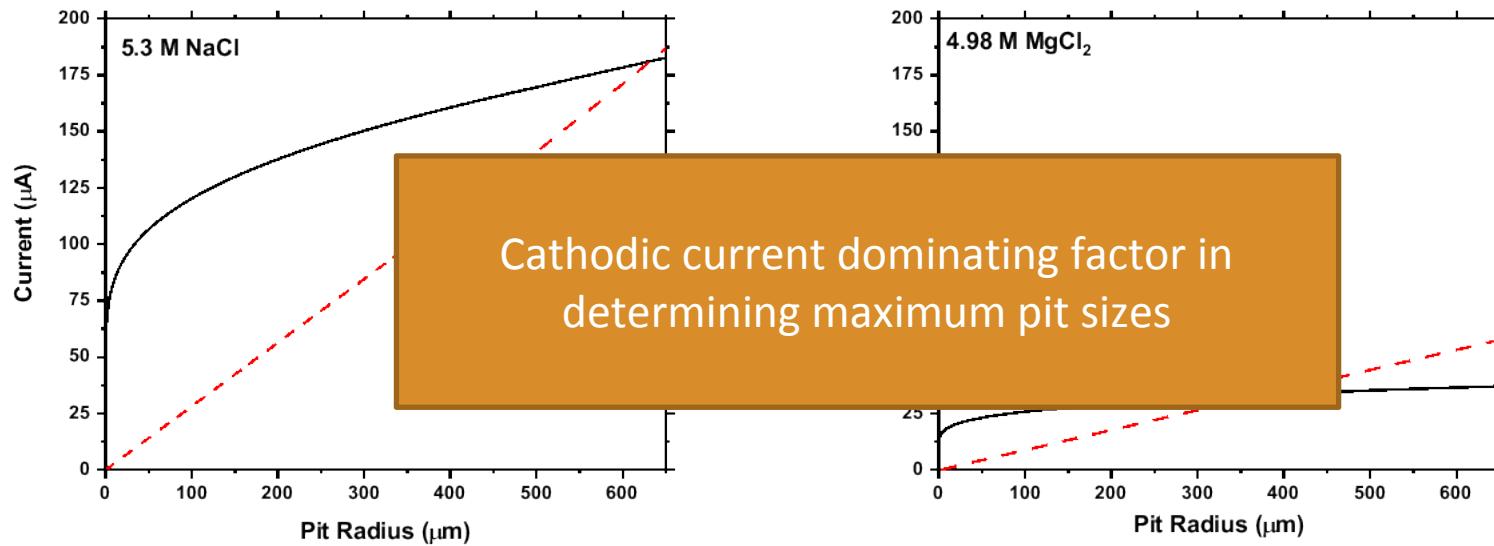
# Elevated Temperatures

- Predicted maximum pit sizes relatively invariant to temperature fluctuations
- Competing phenomena between cathode supply and anodic demand
- Temperature variations may not be as important



# Change in Solution Composition

- Larger pit size in NaCl solutions despite nearly half the chloride concentration



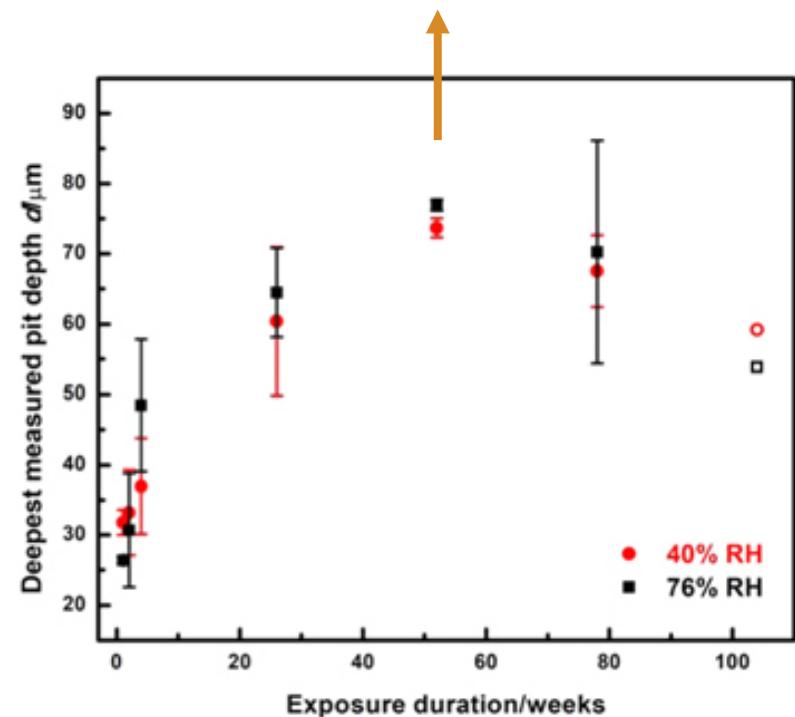
- $\text{MgCl}_2$  experiences ORR suppression and decreased conductivity
- $\text{MgCl}_2$  increased demand for anodic dissolution potentially due to common ion effect

# Comparison to Long Term Exposures

- 104-week exposures of SS304L to sea water solutions at 35 °C
  - 40 % sea salt brine ~ saturated  $\text{MgCl}_2$  with thinner water layer
  - 76 % sea salt brine ~ saturated NaCl
- **Conservative estimates of the maximum pit**
  - Roughly 1.5 x larger estimate

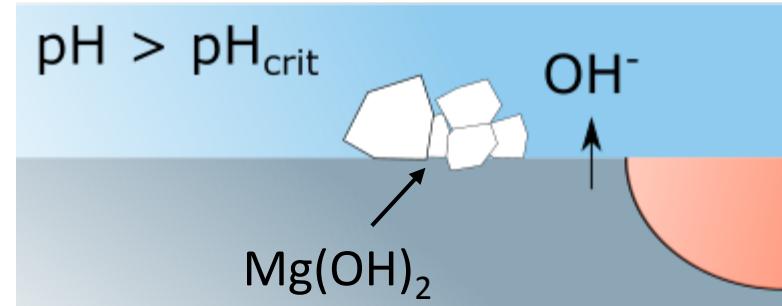
Can we decrease the error in our predicted maximum pit sizes?

76 % Max Pit  $\sim 230 \mu\text{m}$   
40 % Max Pit  $\sim 110 \mu\text{m}$

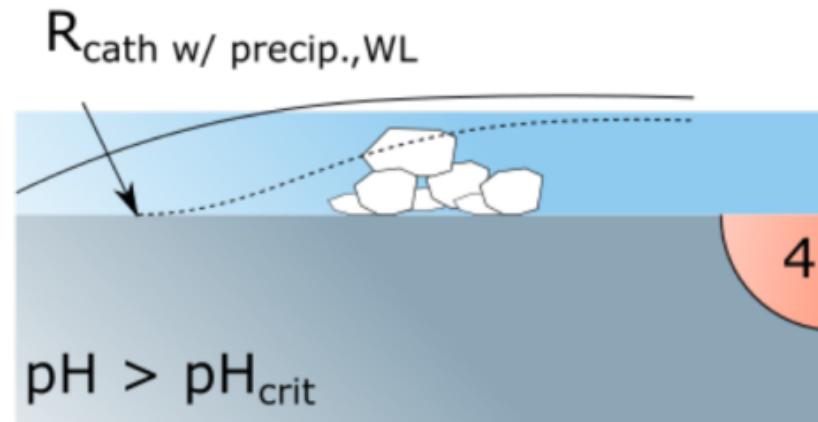


# Improved Prediction for Maximum Pit Sizes

- Current model assumes constant cathode, however, direct observation of precipitates on surface of alloy
- Above  $\text{pH}_{\text{crit}}$  precipitation occurs and can cause:
  - Changes in conductivity
  - Changes in WL thickness
- Will directly impact cathodic kinetics



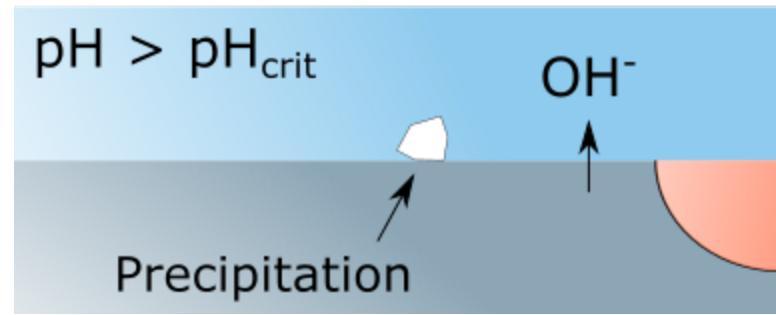
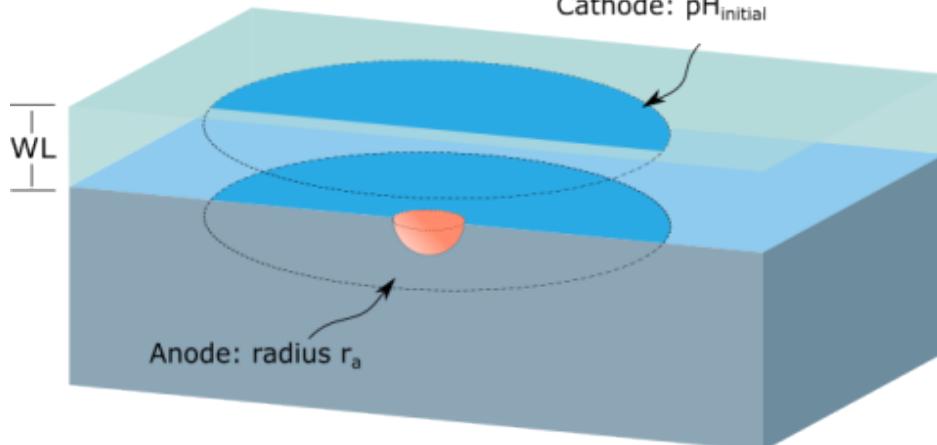
$$I_{c,max} \sim f(\kappa, WL, E_{OCP}, E_{RP}, r_a, i_{eq})$$



# Modeling Cathode Precipitates

- Cathodic reactions cause pH rise
- Can calculate  $\text{OH}^-$  production at a given pit depth and convert to a pH

$$M_{\text{OH}^-} = \frac{\left[ \frac{2}{3} (r_a)^3 \right] \left[ \frac{\rho_{\text{SS}} * e_{\text{SS}}^-}{\text{MW}_{\text{SS}}} \right]}{V_{\text{cath}}}$$



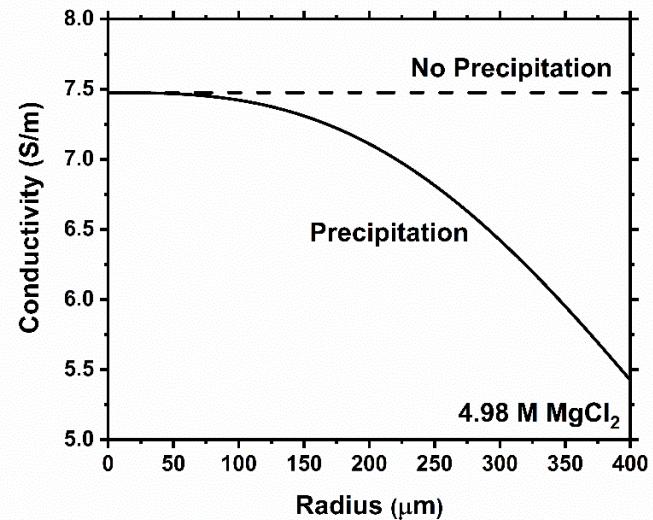
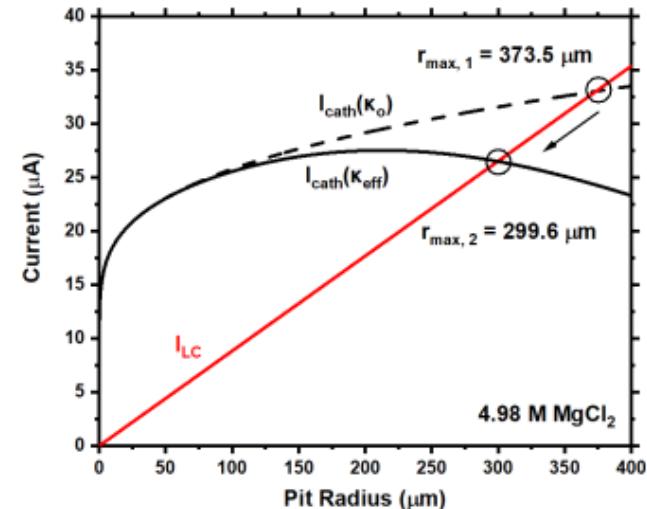
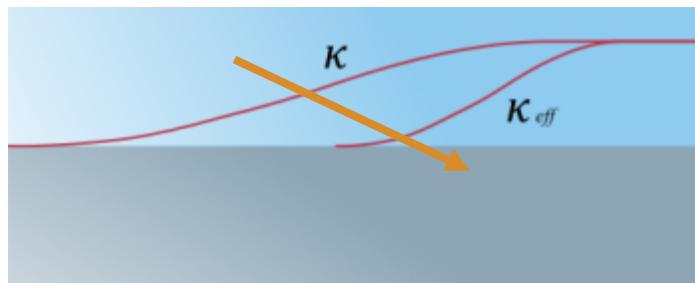
$$\kappa_{\text{eff}} = k \left( 1 - \frac{V_{\text{precip}}}{V_{\text{solution}} + V_{\text{precip}}} \right)^{\frac{3}{2}}$$



$$I_{\text{cath,updated}} = f(\kappa_{\text{eff}})$$

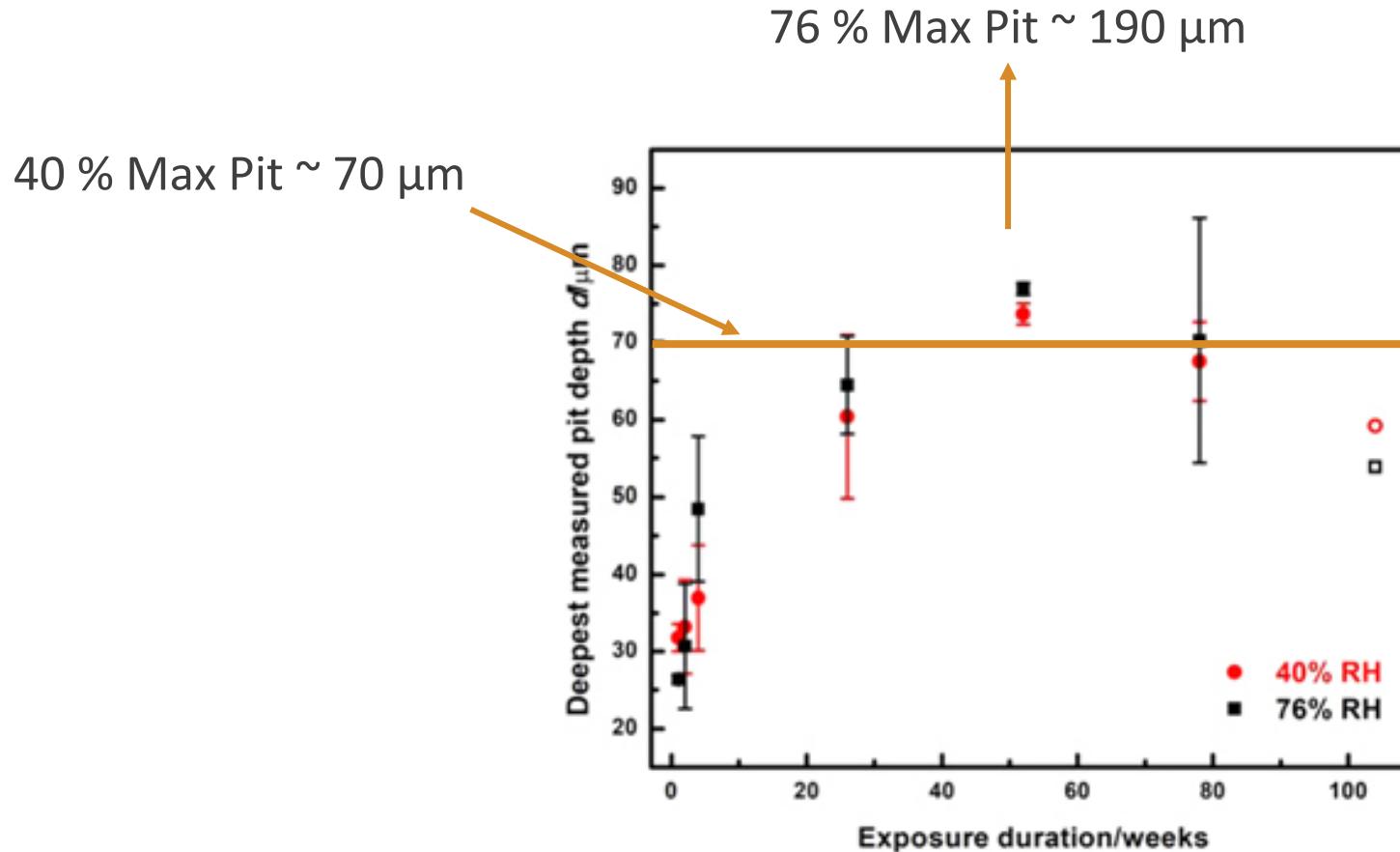
# Influence of Precipitation on Max Pit Sizes

- Initially for  $\text{MgCl}_2$  solutions
- Decrease in pit size by roughly 75  $\mu\text{m}$  when considering precipitation
- Decrease in conductivity by roughly a factor of 1.5



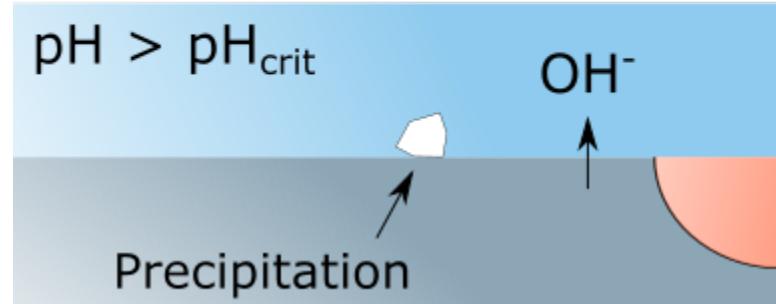
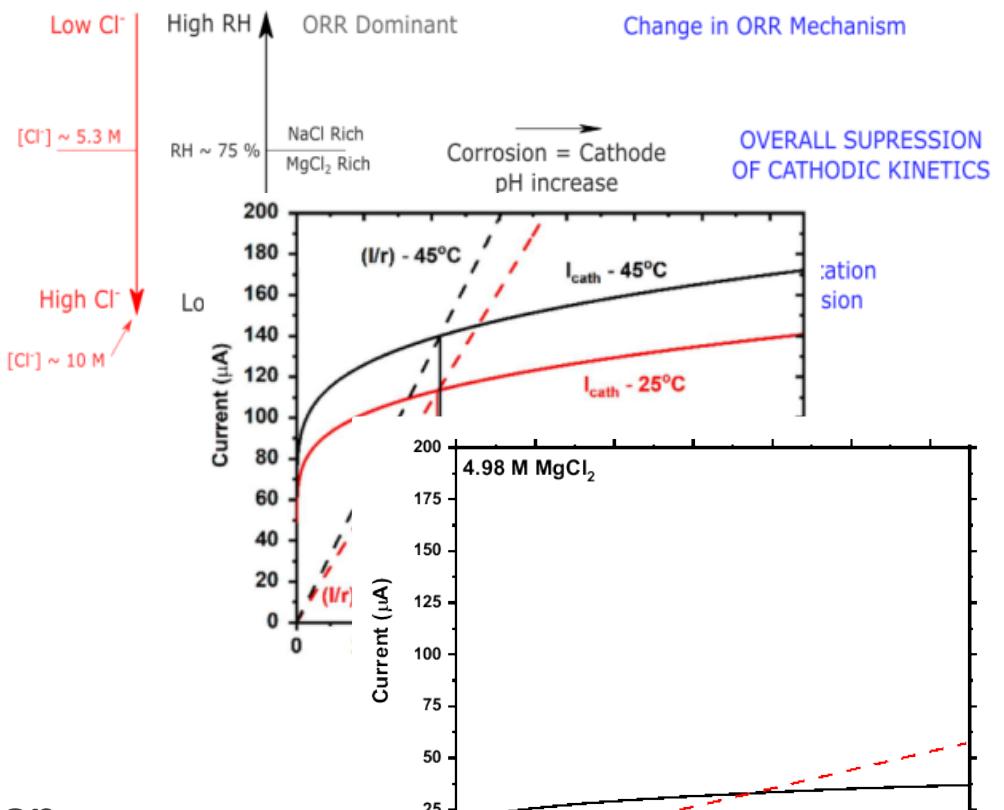
# Re-Comparison to Long Term Exposures

- When comparing to exposures, prediction of maximum pit sizes with precipitation is directly inline for 40% RH



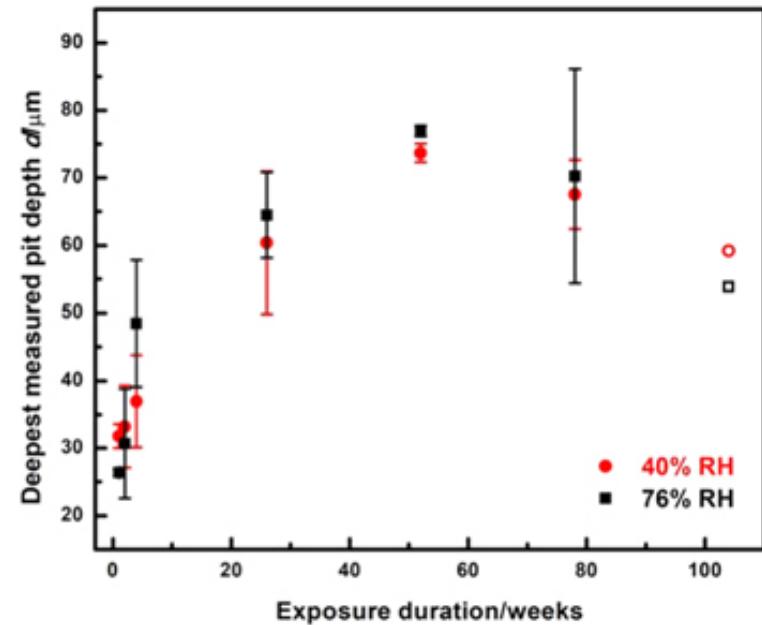
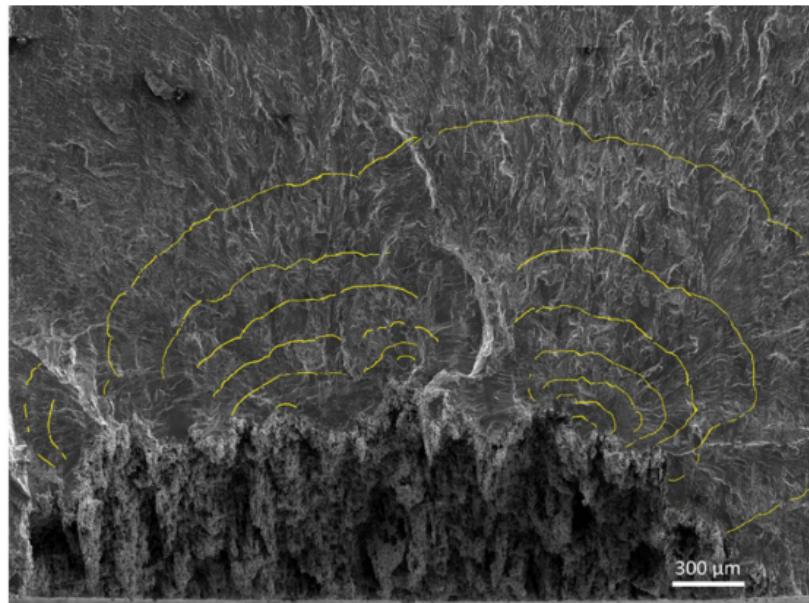
# Answered Questions

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- To what extent can we predict damage from localized corrosion and stress corrosion cracking and what are the governing factors in these predictions?



# What is still missing?

- Kinetics of pit growth and metastable pitting
- Pit to crack transition



- Stress Corrosion Cracking
  - In-situ crack growth rate measurements in atmospheric conditions
  - Understanding crack tip electrochemistry

# Extension of Work, Future Work, and Potential Applications

# Extension of Work to Galvanic Modeling

- Finite Element Modeling in a galvanic couple
- Many trends with temperature and chloride concentration hold in galvanic couple



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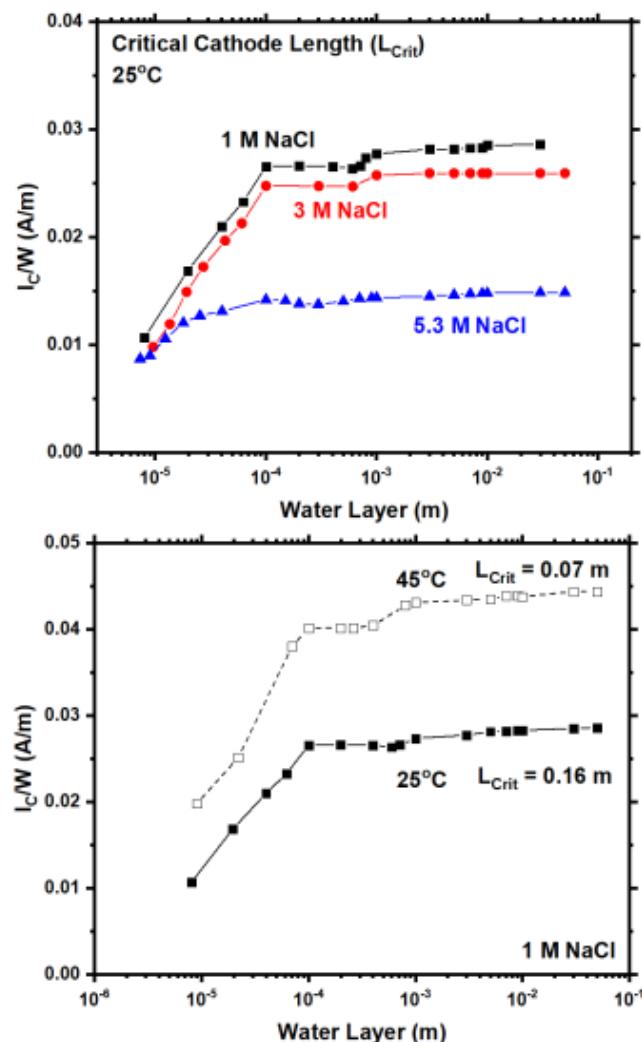
**Editors' Choice—Natural Convection Boundary Layer Thickness at Elevated Chloride Concentrations and Temperatures and the Effects on a Galvanic Couple**

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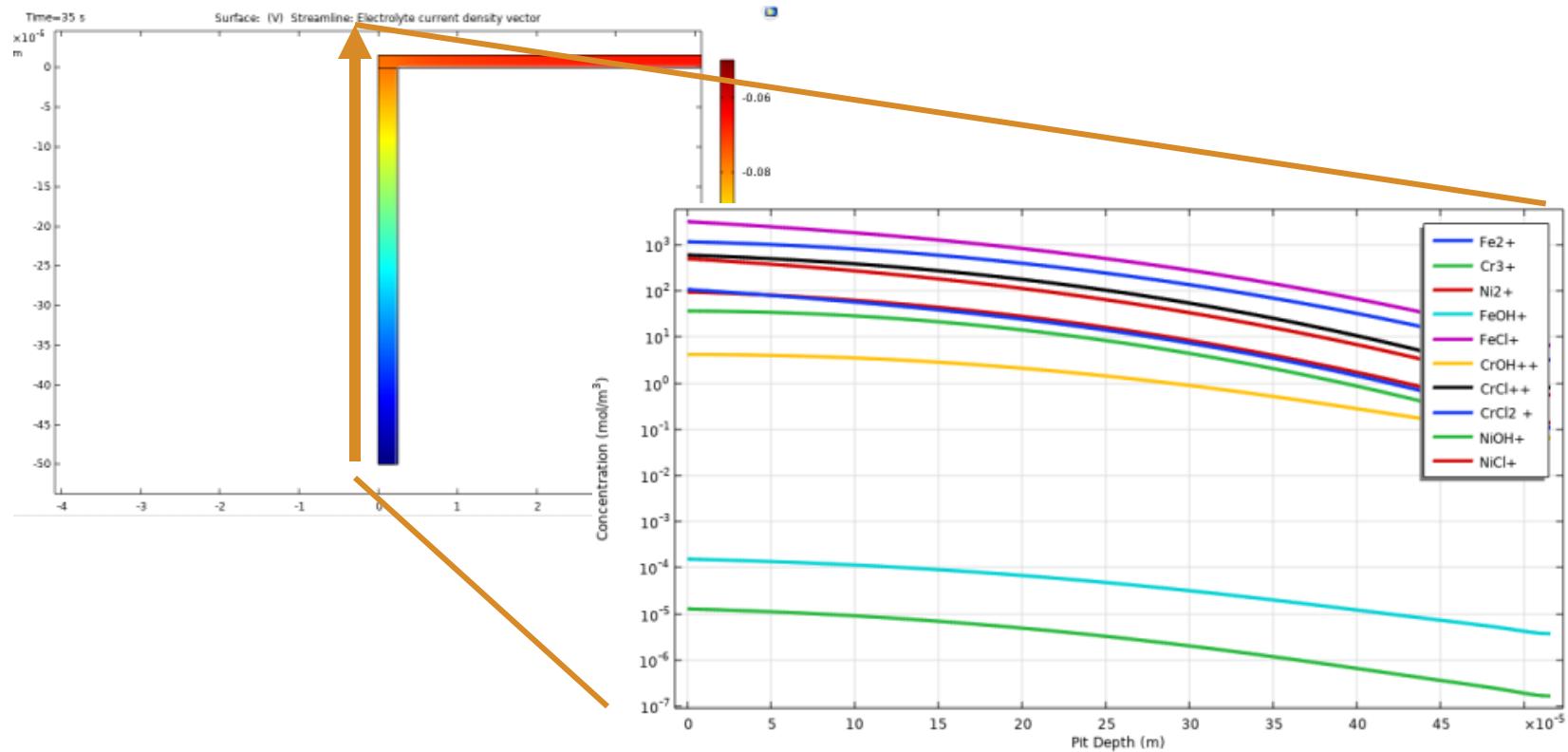
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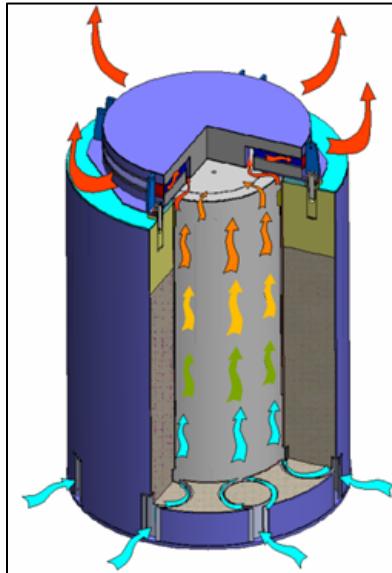
# Future work

- Enhanced modeling tracking pH and chemical speciation using finite element modeling techniques

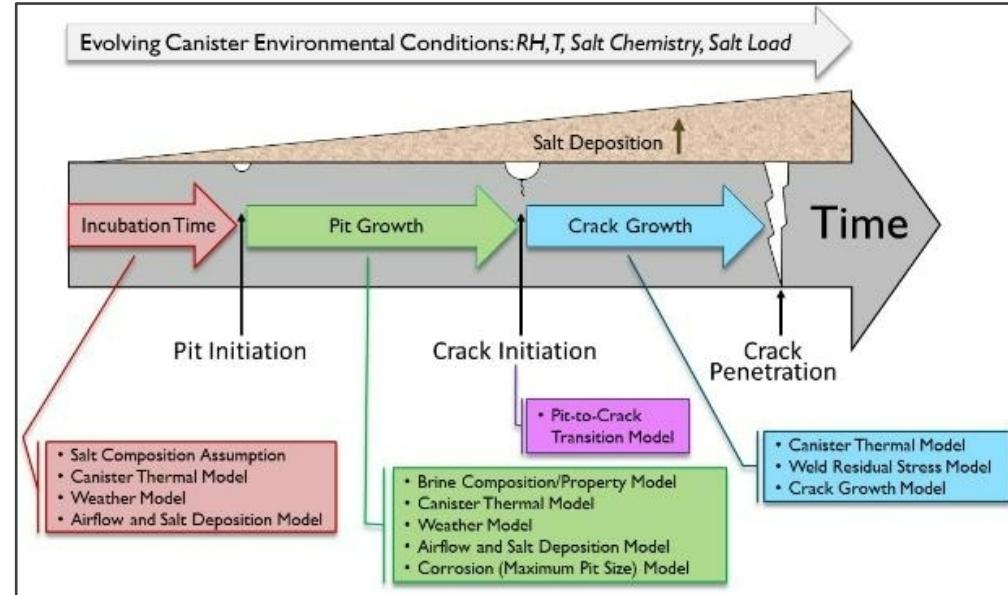


# Modeling in Consulting

- Overall project was supporting modeling efforts to help identify potential materials degradation for spent nuclear fuel canisters
- Goal to help identify potential hot spots for materials degradation (local to canister and around the country)

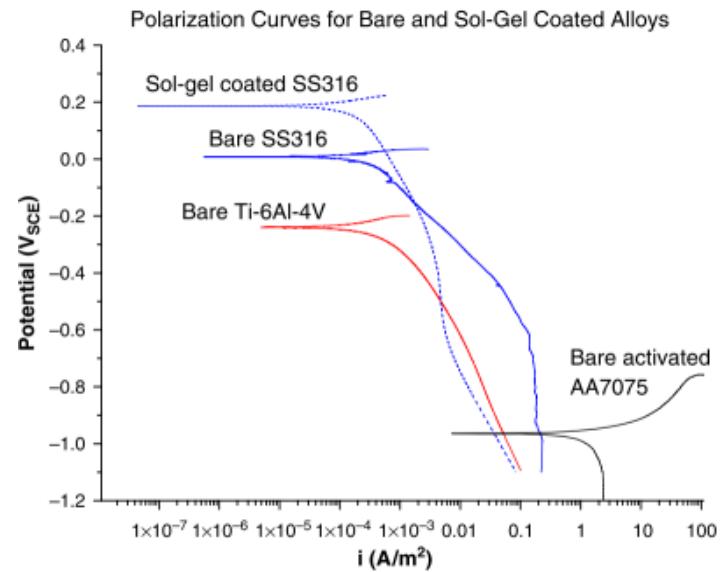
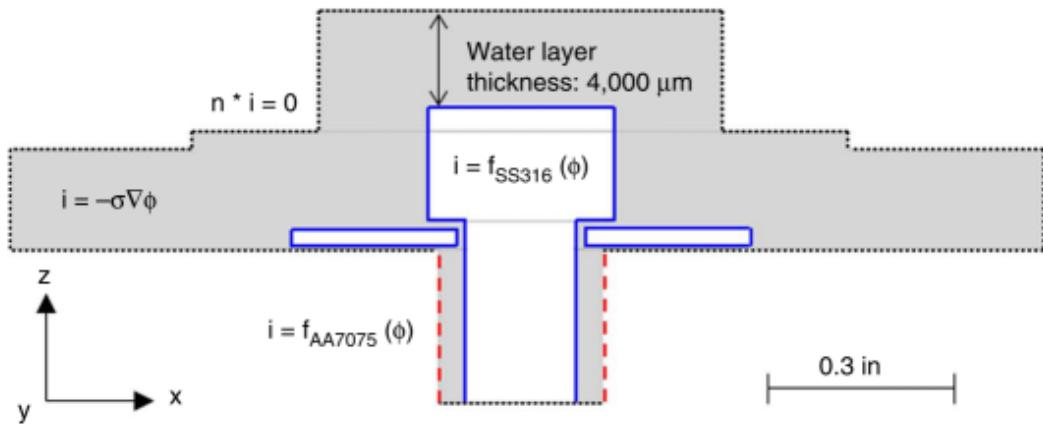


FCRD-UFD-2012-000114 Figure 7.3



# Modeling in Consulting

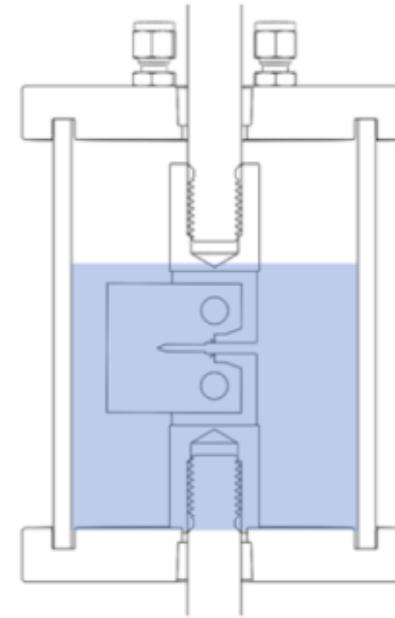
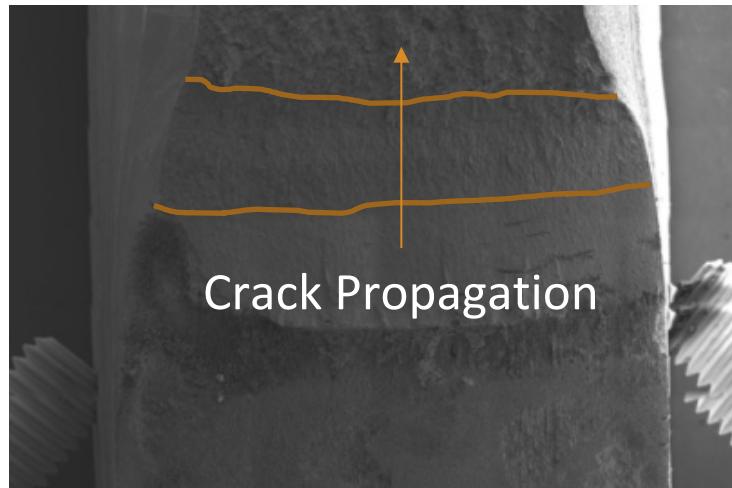
- Utilizing Finite Element Modeling in order to inform upon coating efficacy and potential failure



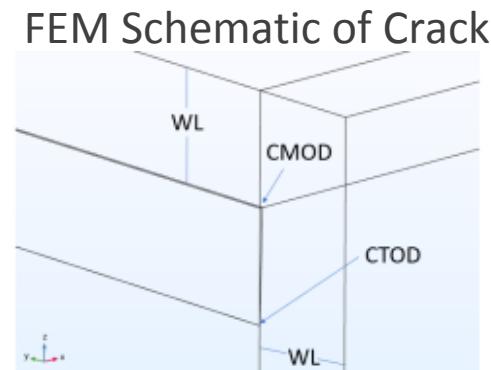
- Identify potential areas for failure in different geometries

# Failure Analysis

- Assessing failure mechanisms by *in-situ* crack growth rate
- Performing ex-situ microscopy in order to help determine stress corrosion cracking morphologies



- Combine with modeling to get overall picture of cracking mechanisms



# Acknowledgements

- Robert Kelly, Kelly Research Group, Center for Electrochemical Science and Engineering
- Rebecca Schaller, Charles Bryan, Sylvia Saltzstein, Jacob Carpenter, Andrew Knight, Eric Schindelholz



# Understanding Governing Factors of Localized Corrosion in Marine Environments

R. M. Katona<sup>1,2</sup>

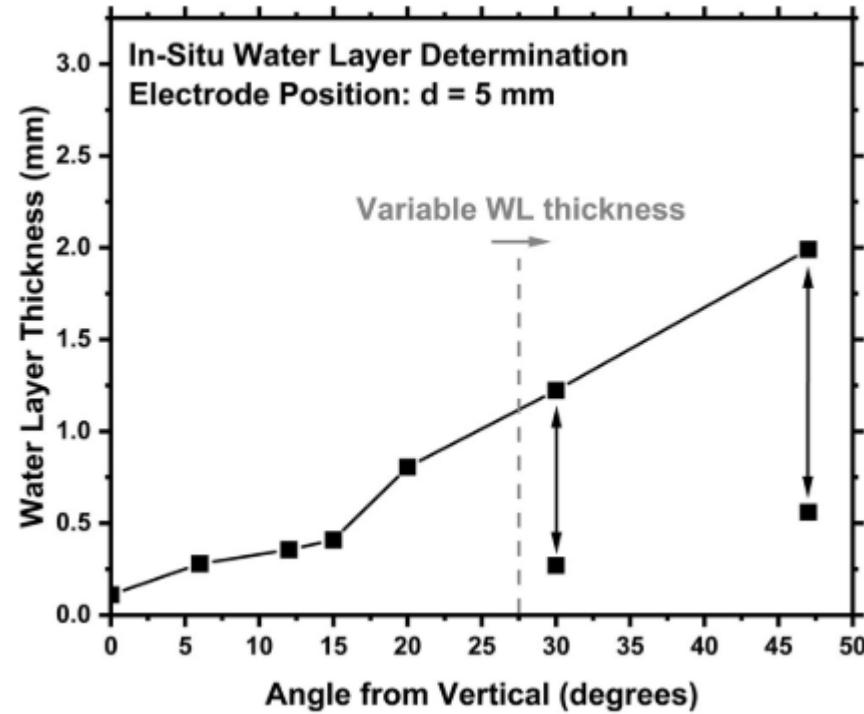
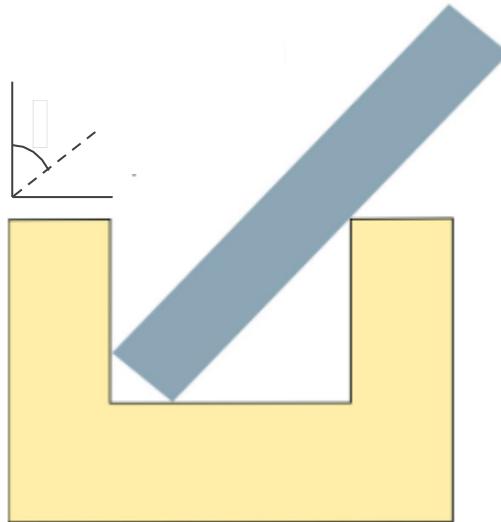
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# Supplemental

# Putting the WL into Perspective

- In accelerated corrosion scenarios, WL thicknesses are based on angle of exposure



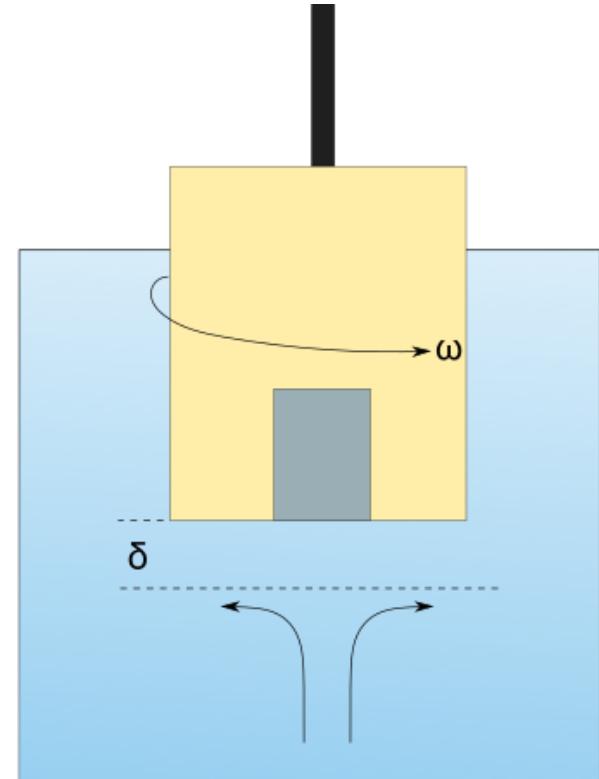
- Under ASTM B117 (15 – 30°), WL varies from 660 – 1210  $\mu\text{m}$
- Transient WL present at certain angles of exposure

# Putting the WL into Perspective

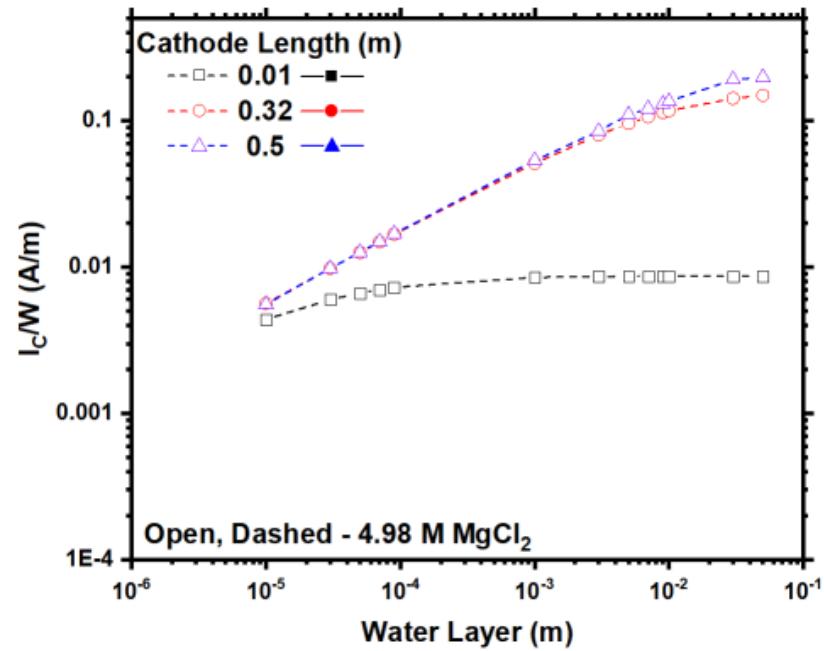
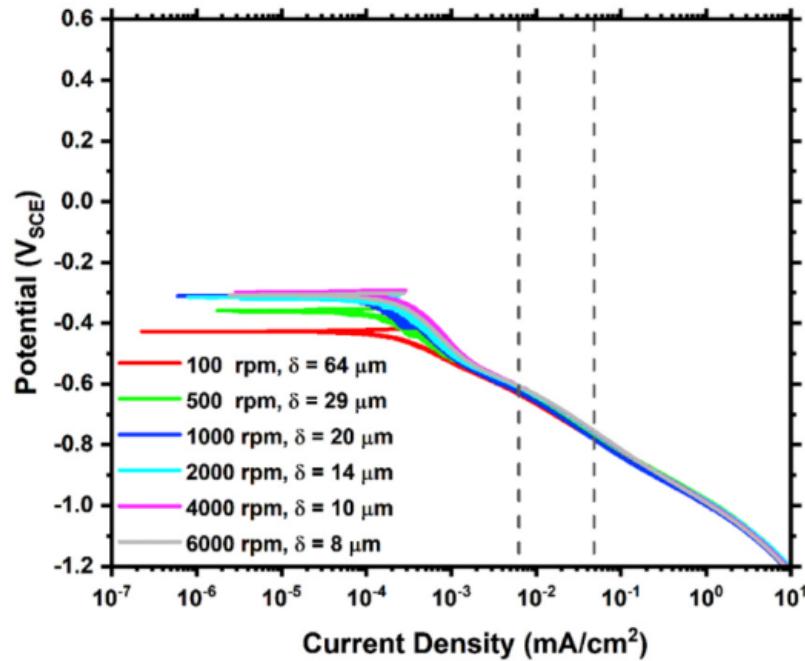
- When using a rotating disk electrode (RDE) a hydrodynamic boundary layer can simulate WL thicknesses

$$\delta \propto \frac{D^{1/3} \nu^{1/6}}{\omega^{1/2}}$$

- Dependent upon diffusion coefficient ( $D$ ), kinematic viscosity ( $\nu$ ) and the rotation rate ( $\omega$ )
- Values range from  $\sim 7$  to  $200 \mu\text{m}$  depending on solution values and rotation rate

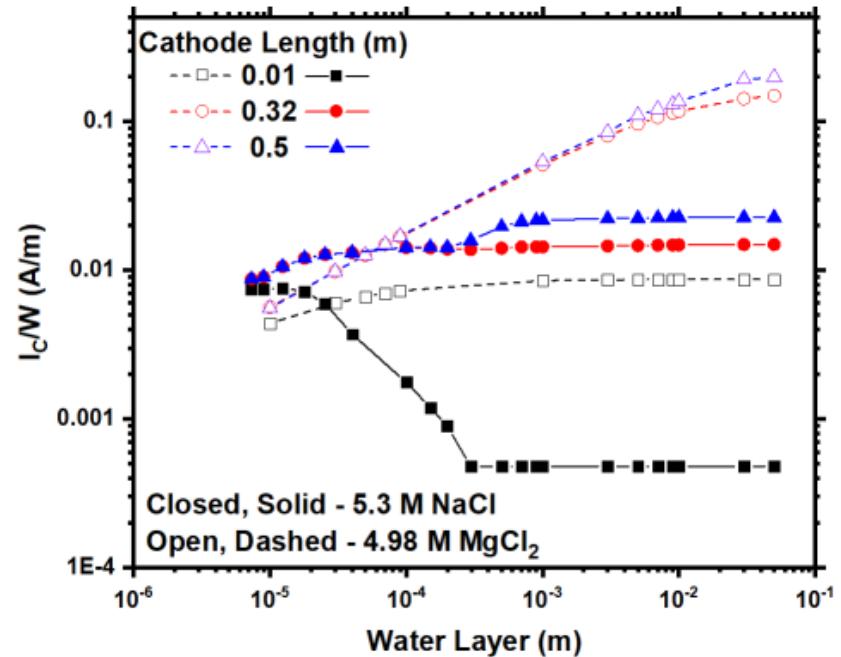
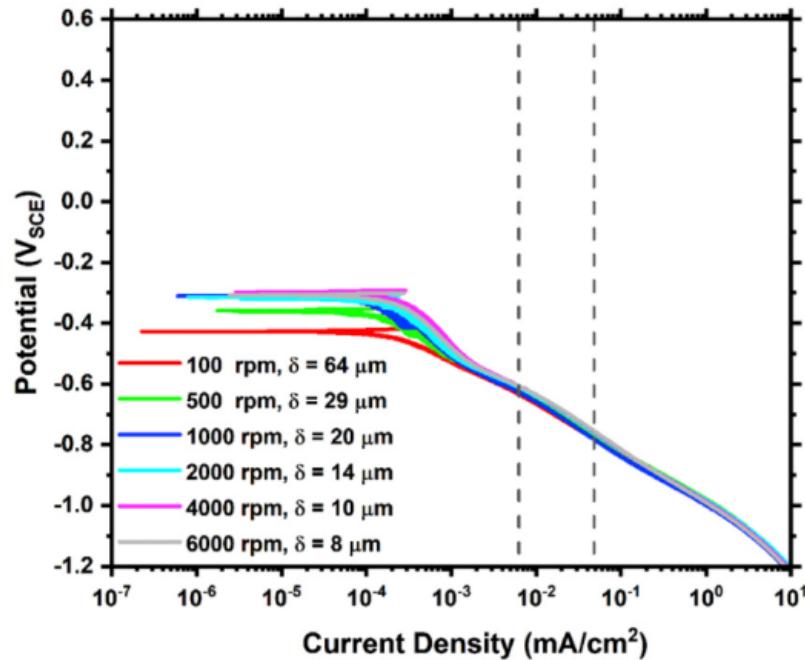


# Changing Solution



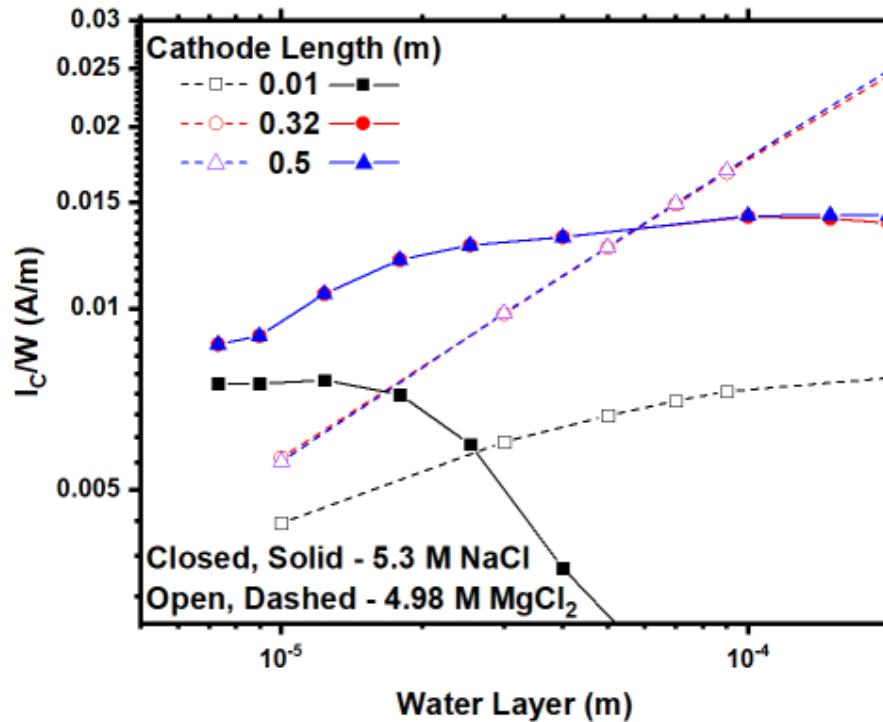
- MgCl<sub>2</sub> solutions exhibit no rotational dependence when using an RDE
- I<sub>C</sub>/W only increases with increasing WL
  - Ohmic control
  - No M-T limitations

# Changing Solution



- $\text{MgCl}_2$  solutions exhibit no rotational dependence when using an RDE
- $I_C/W$  only increases with increasing WL
  - Ohmic control
  - No M-T limitations

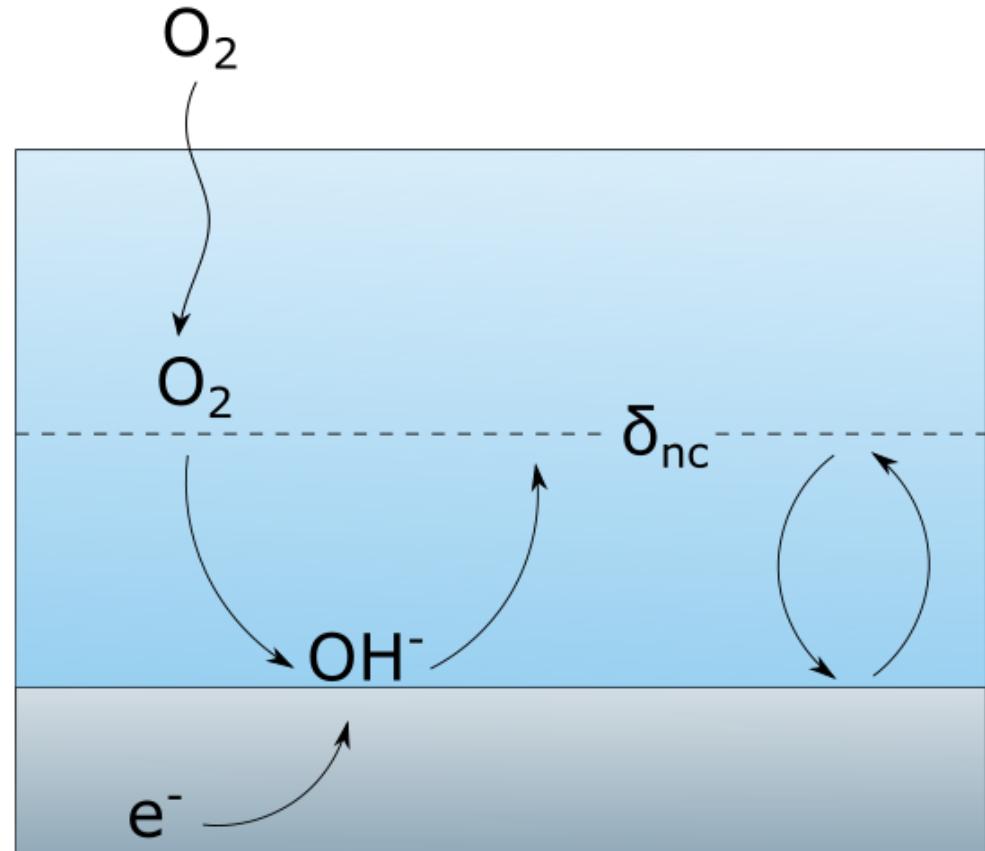
# Changing Solution



- At small WL, NaCl has a higher  $I_C/W$ 
  - Due to current density dependence on WL
- When increasing WL, NaCl experiences more M-T limitations whereas  $MgCl_2$  does not due to a lack of WL dependence on the cathodic kinetics

# Natural Convection Boundary Layer

- Natural convection occurs in solutions that have thermally or compositionally driven spatially inhomogeneous density distributions
- When oxygen is being reduced:



- Determines the transition between atmospheric and bulk conditions

# Natural Convection Boundary Layer

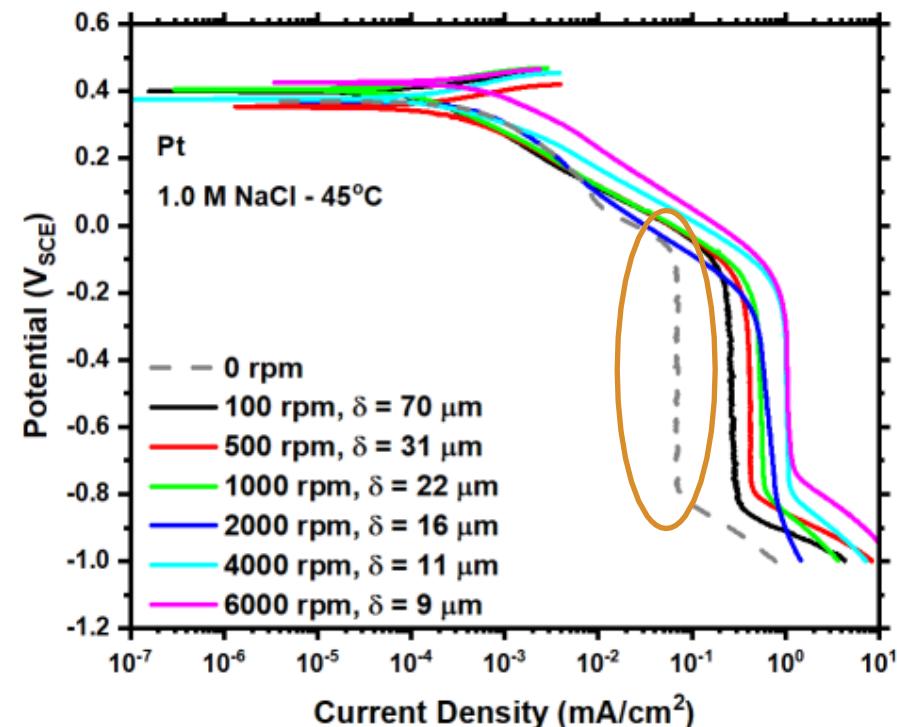
- Limiting current densities can be predicted based on Fick's first law:

$$i_{lim} = \frac{nFD(C_{bulk} - C_{surface})}{\delta}$$

- If there was no natural convection creating a boundary layer,

$$\delta \rightarrow \infty$$

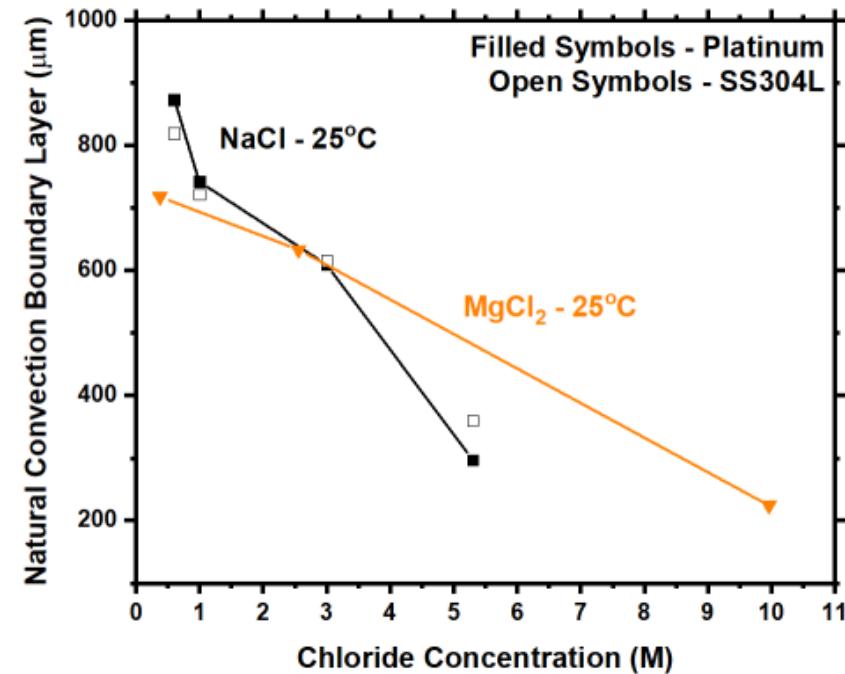
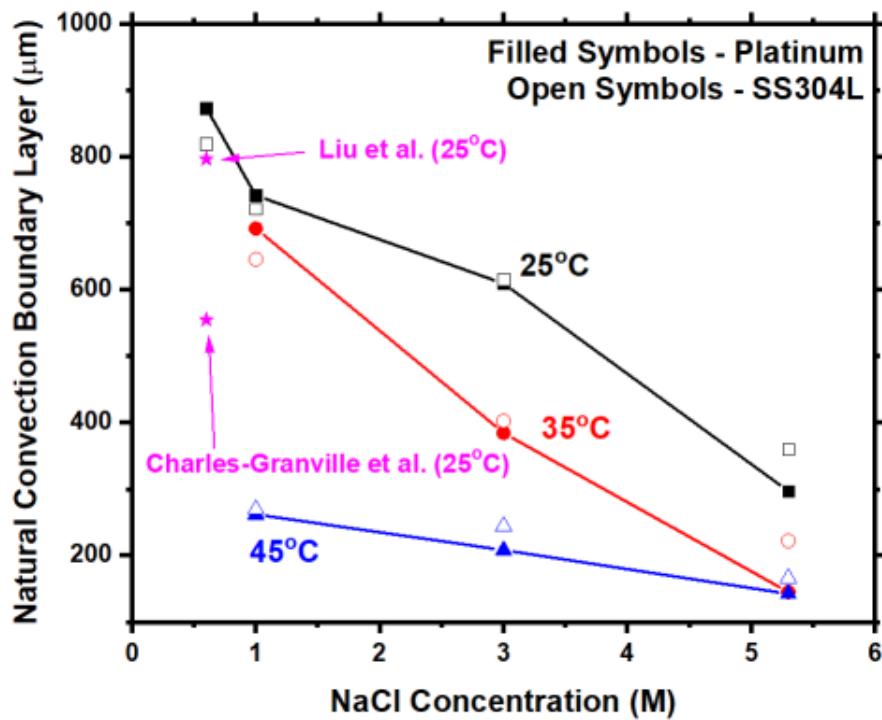
$$\begin{array}{c} \text{I} \\ \text{I} \\ \text{I} \end{array} \rightarrow 0$$



- Liu et al. determined the natural convection boundary layer to be roughly 800  $\mu\text{m}$  in 0.6 M NaCl

# Natural Convection Boundary Layer

- Increasing NaCl concentration decreases  $\delta_{nc}$
- Increasing temperature decreases  $\delta_{nc}$



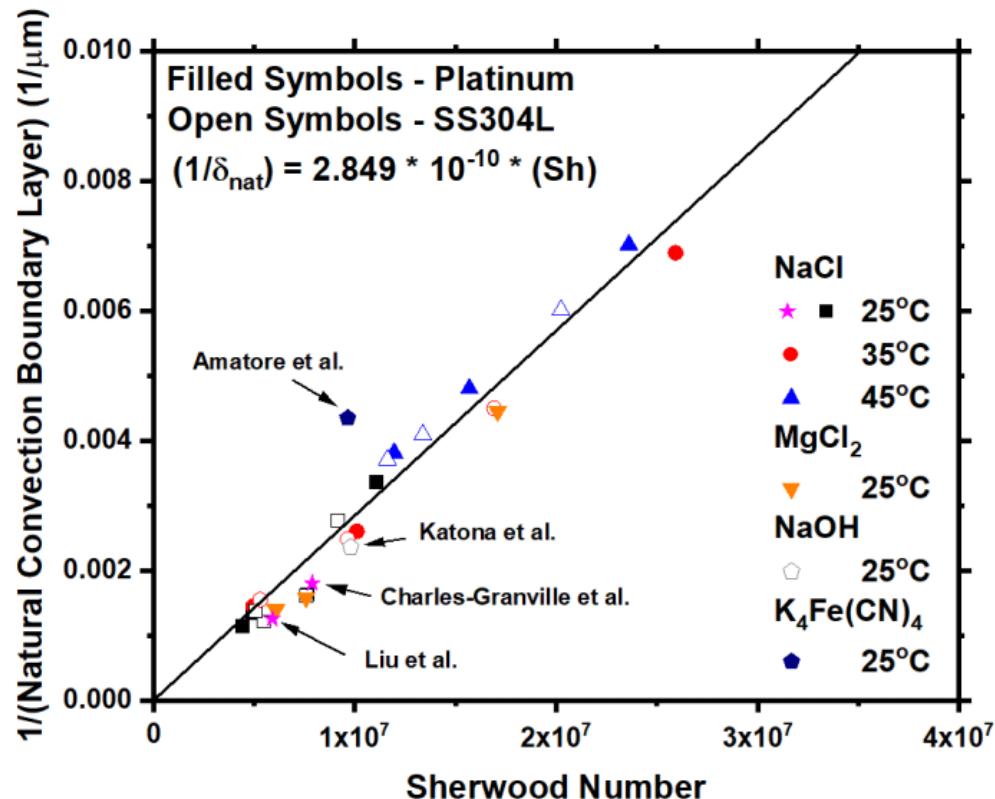
- Trends hold for  $\text{MgCl}_2$  solutions

# Can we predict $\delta_{nc}$ ?

- $\delta_{nc}$  does not scale with  $i_{lim}$  or any other singular solution property ( $C_{\infty}$ ,  $D_{\infty}$ , etc.)
- Mass transport in electrochemical systems can be described with dimensionless values

$$K = \frac{i_{lim}}{nFC_{bulk}}$$

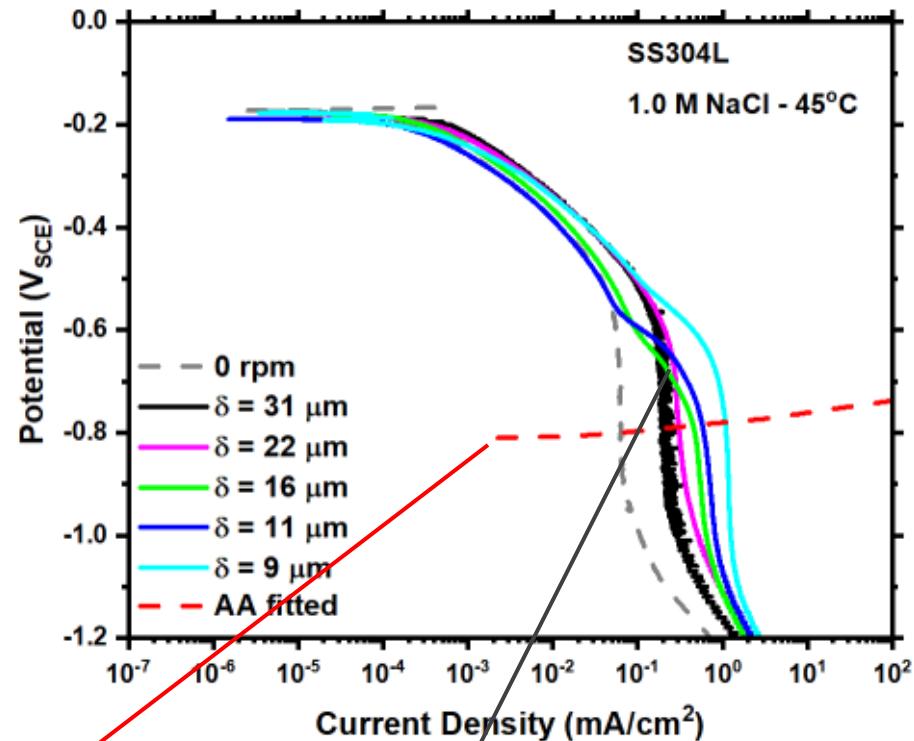
$$Sh = \frac{K}{D/d}$$



- Literature results match well with fit

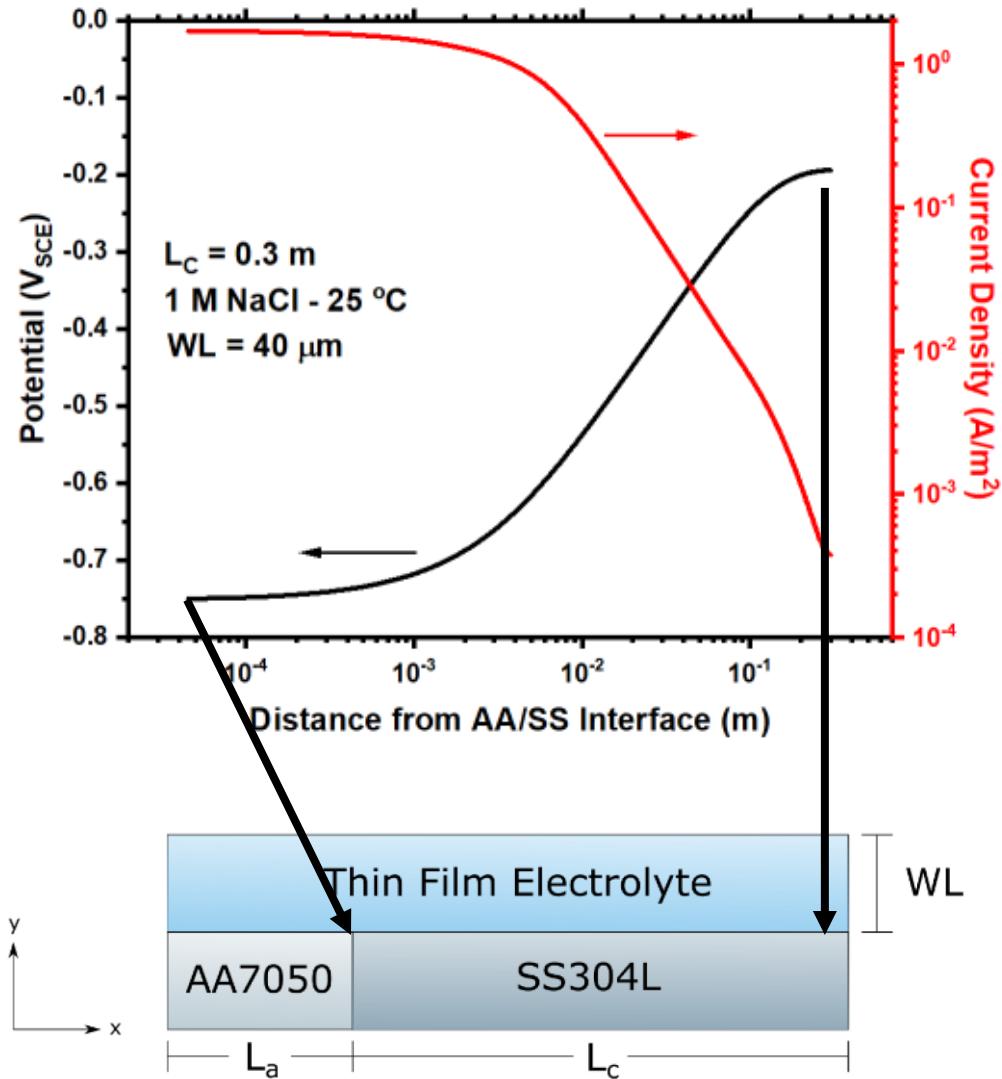
# Modeling in a Galvanic Couple

- Assign the polarization scan to respective anode or cathode
  - Anything greater than  $\delta_{nc}$  utilizes quiescent polarization scan
- Input solution conductivity
- Variable cathode length ( $L_c$ ) and WL thickness
- Evaluate current per width ( $I_c/W$ )



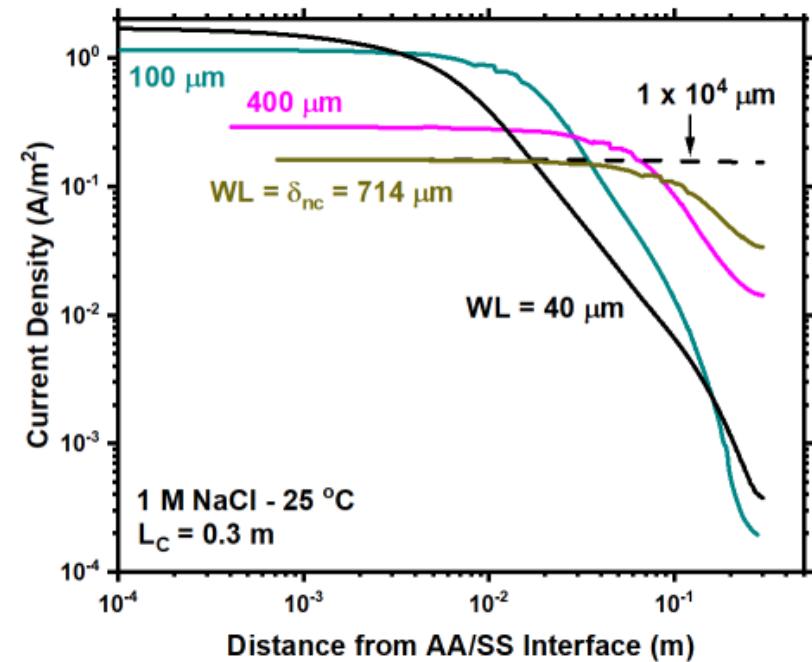
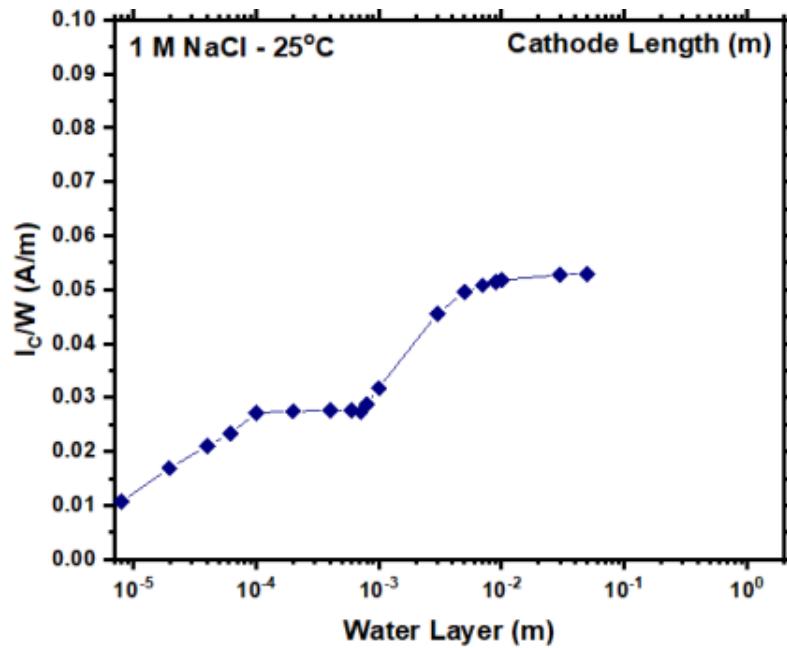
# Modeling in a Galvanic Couple

- Potential at the interface is the AA/SS304L coupling potential
- Potential far away from interface is OCP of SS304L
- Current density high near the interface and decays when moving from the interface
- Evaluate  $I_c/W$  by integrating current along the cathode



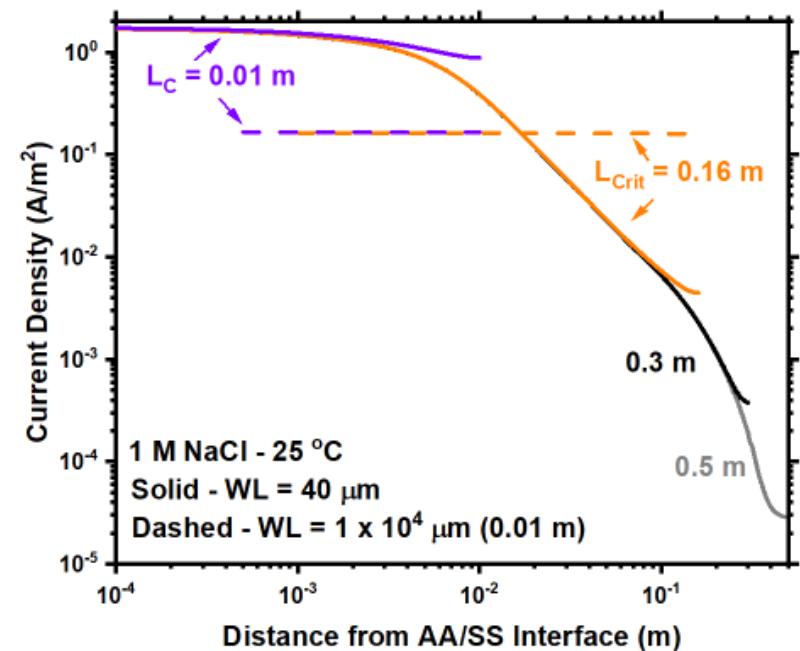
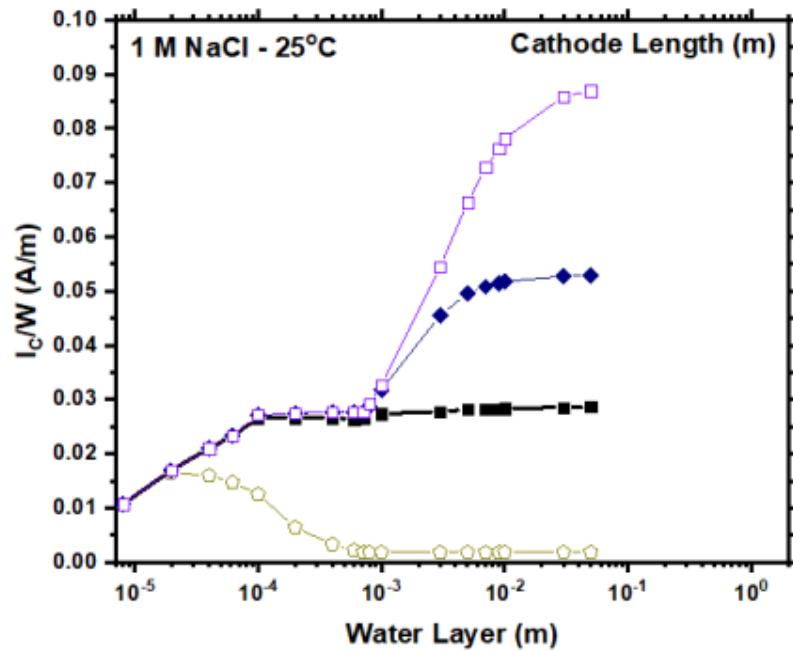
# Increasing WL Thickness

- Continuously increasing the WL at the same cathode size will not increase the current
- Current is now fully limited by the cathode size

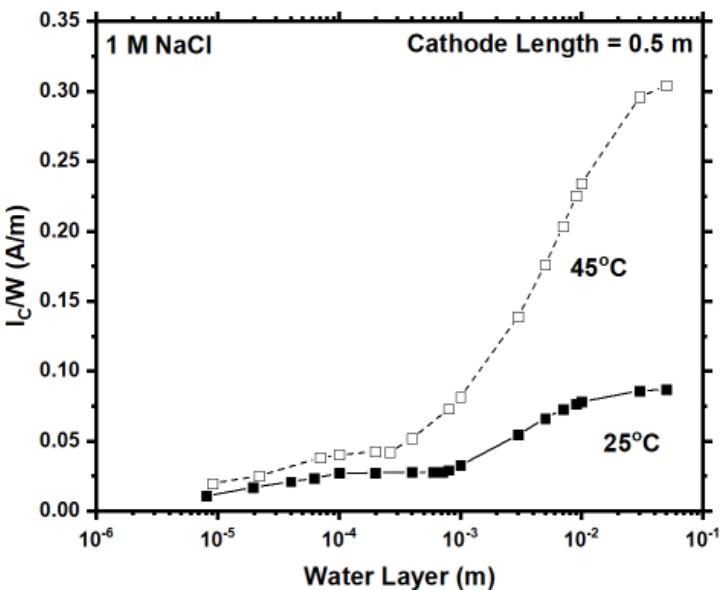
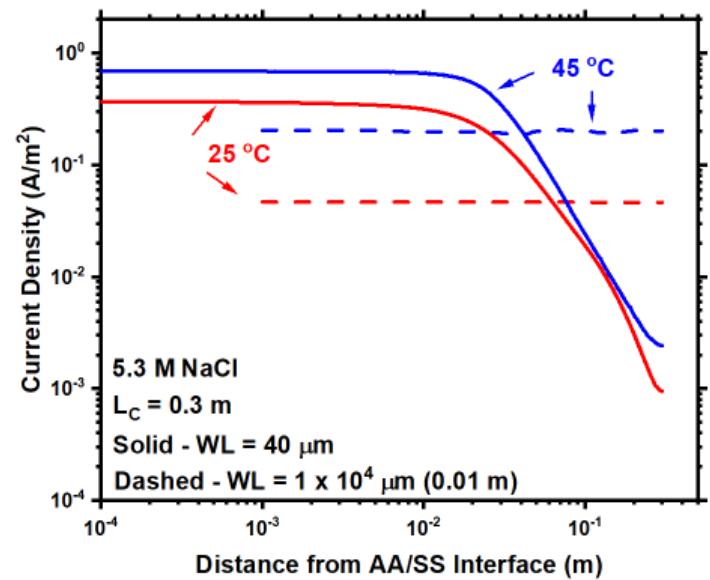
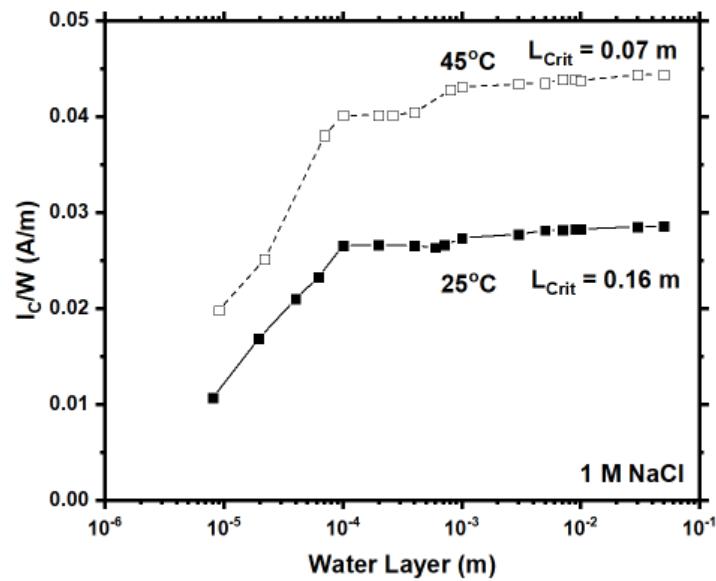


# Increasing Cathode Length and WL

- Further increasing cathode length increases  $I_C/W$
- $I_C/W$  scales with cathode size

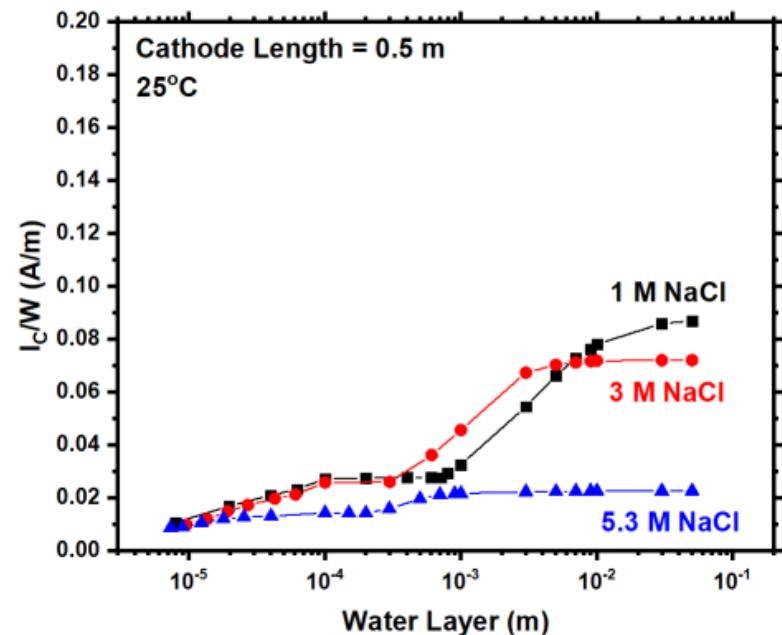
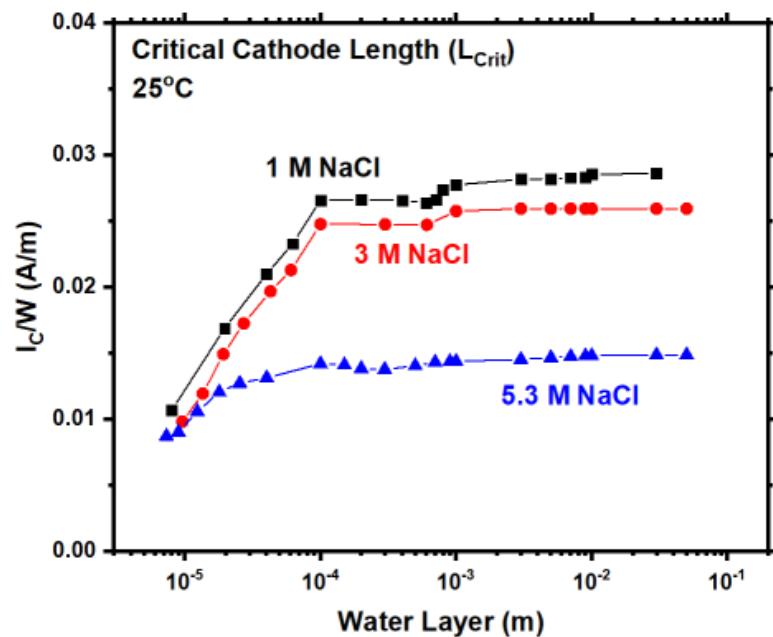


# Increasing Temperature



- Increasing temperature increases  $I_C/W$
- Increasing temperature increases  $\kappa$  and  $i_{\text{lim}}$

# Increasing Chloride Concentration

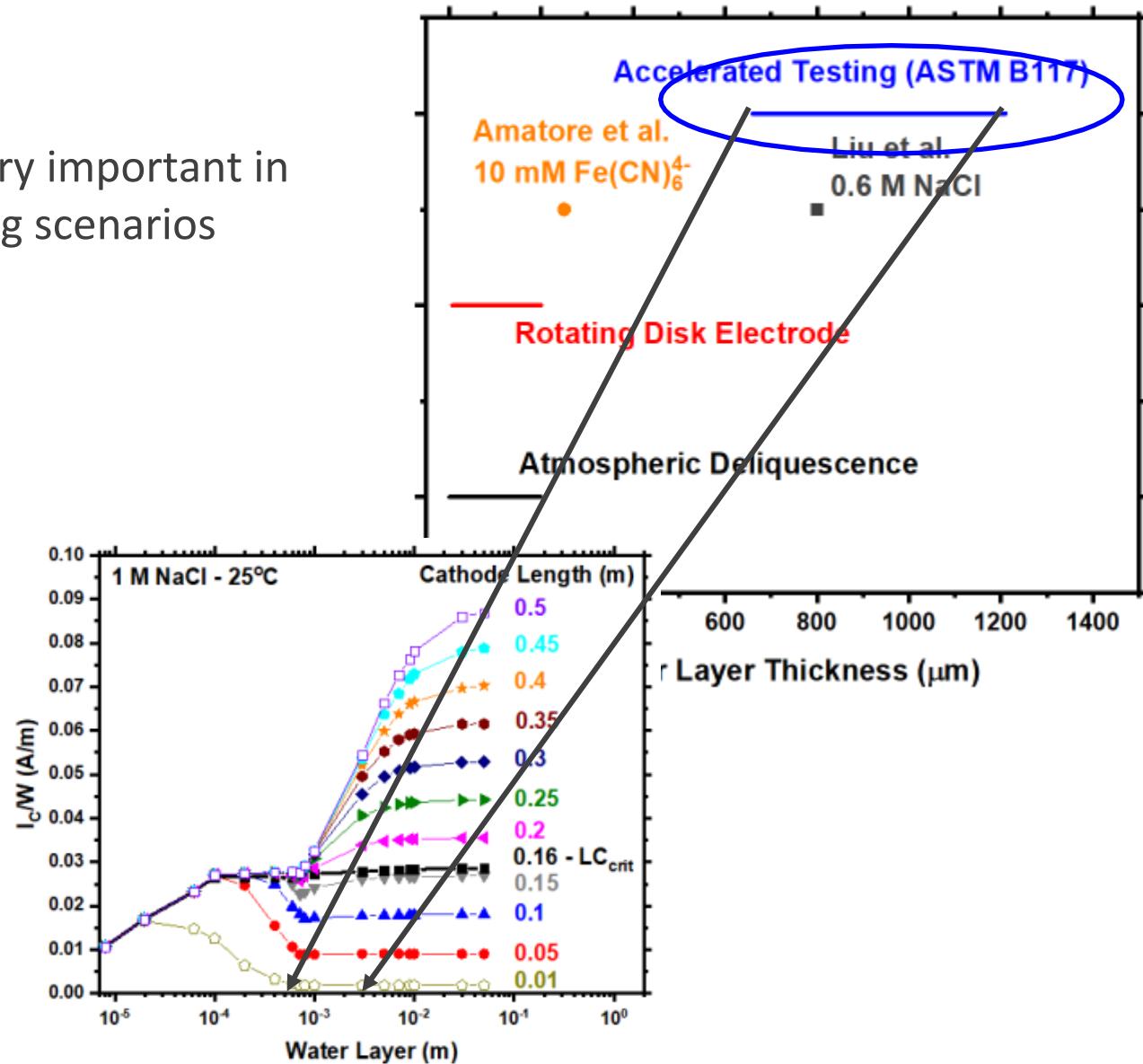


- Increasing chloride concentration decreases  $I_c/W$  in most cases
- Shouldn't current increase?  $\rightarrow$  More corrosion damage?
- Limited by the mass transport in the system

$$i_{lim} = 0.62nFD_{O_2}^{2/3}v^{-1/6}C_{O_2,bulk}\omega^{1/2}$$

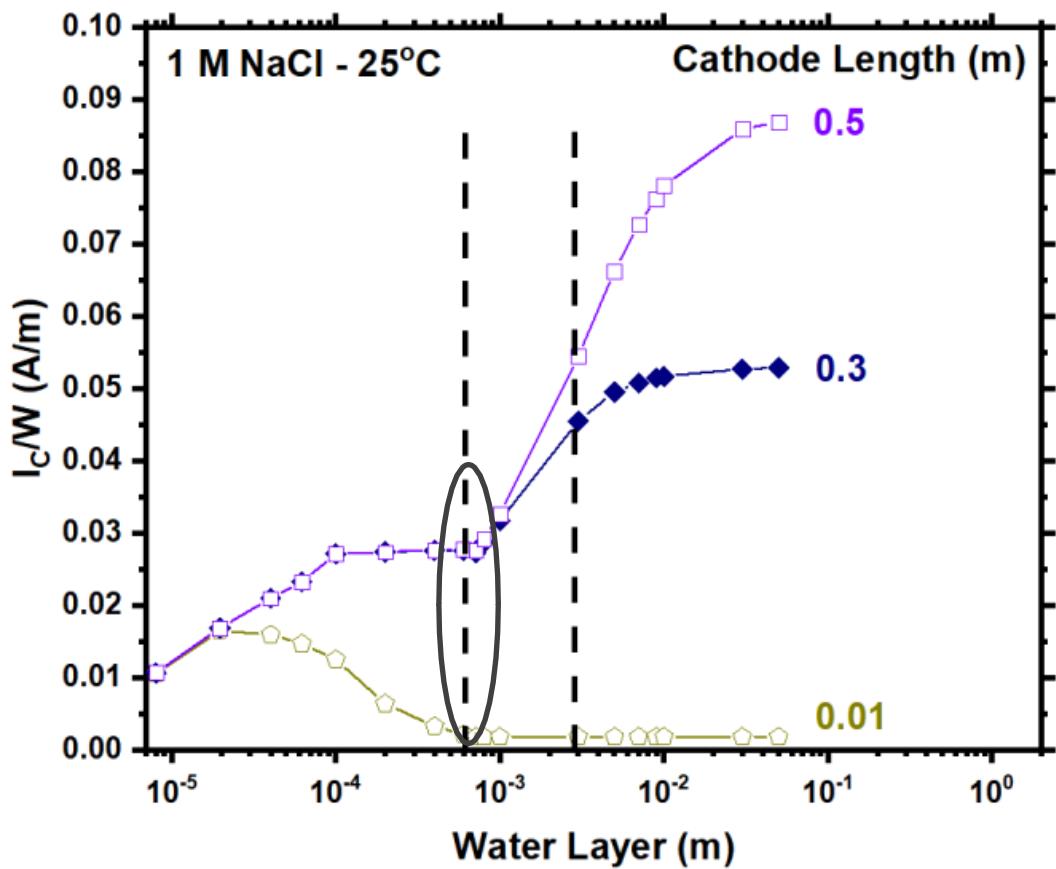
# Implications

- Cathode size is very important in accelerated testing scenarios



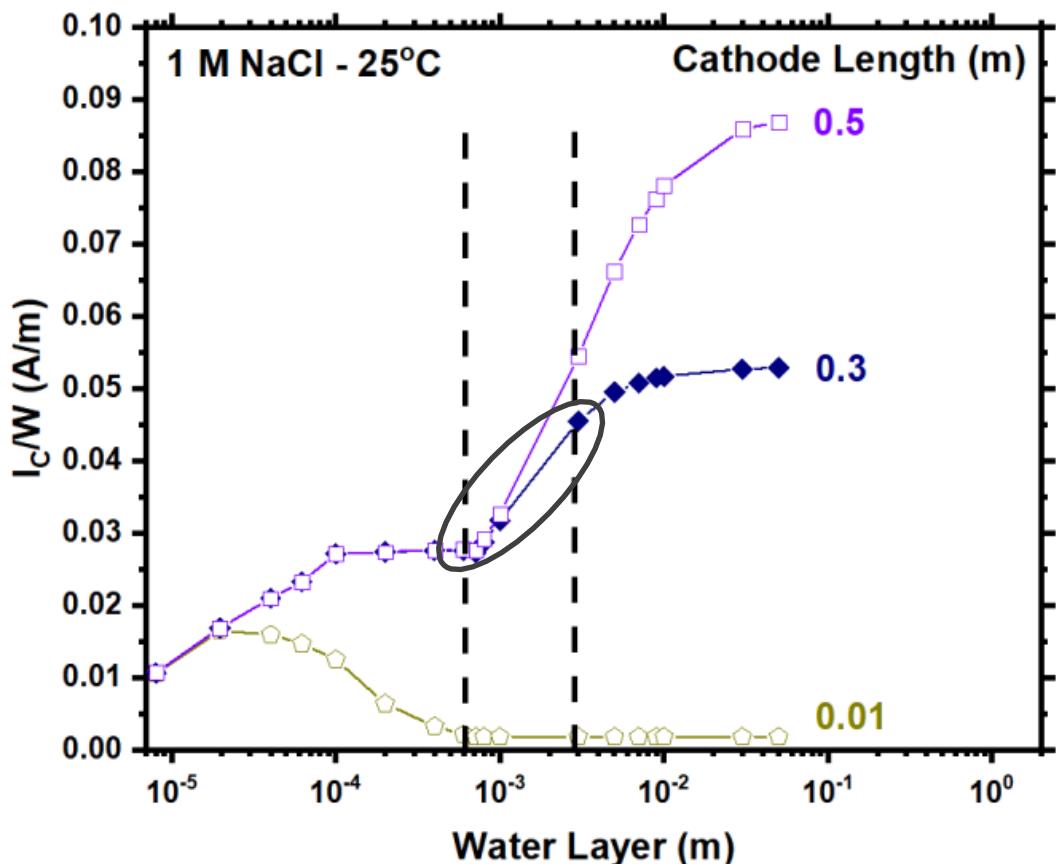
# Implications

- Cathode size is very important in accelerated testing scenarios
- Using small cathodes severely underestimates corrosion damage
  - $I_c/W$  is 15 times greater at the lower angle for the larger cathodes (0.3 and 0.5 m) in comparison to the smaller cathode (0.01 m)



# Implications

- Cathode size is very important in accelerated testing scenarios
- Using small cathodes severely underestimates corrosion damage
  - $I_c/W$  is 15 times greater at the lower angle for the larger cathodes (0.3 and 0.5 m) in comparison to the smaller cathode (0.01 m)
- Same cathode length at different angles will experience different corrosion damage
  - Tests experience large test-to-test and chamber-to-chamber variability
  - Difficult to extrapolate corrosion damage to real life scenarios



# MgCl<sub>2</sub> Cathodic Kinetics

