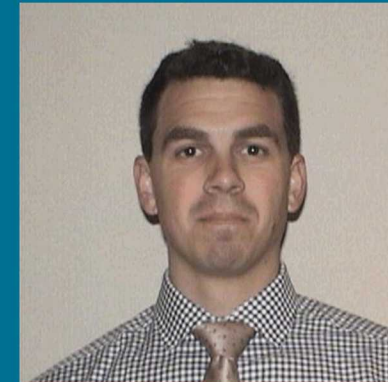
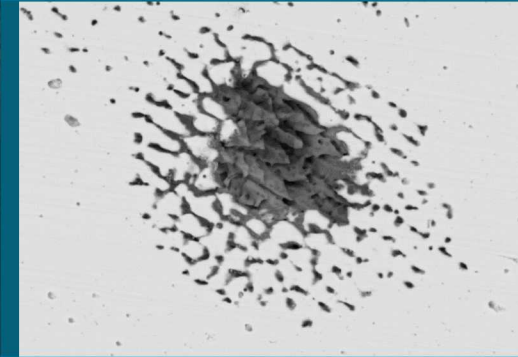
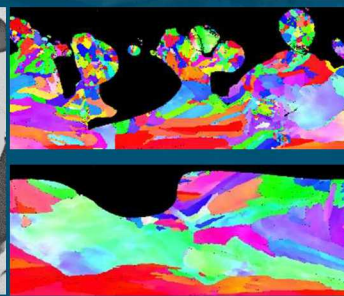
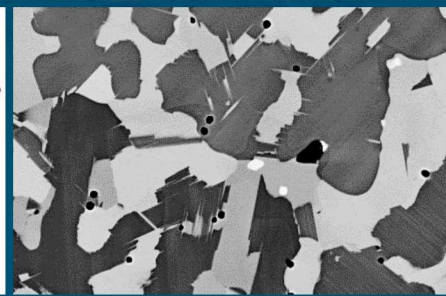
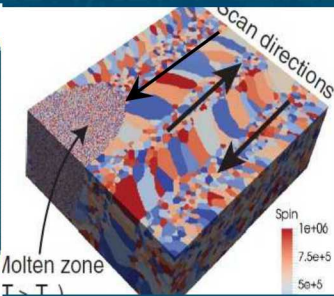
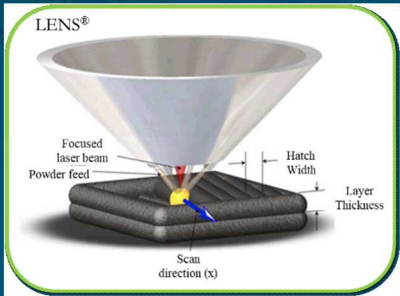


# Additive Manufacturing to Compositionally Grade Metals for High Throughput Alloy Screening



Michael Melia (SNL)

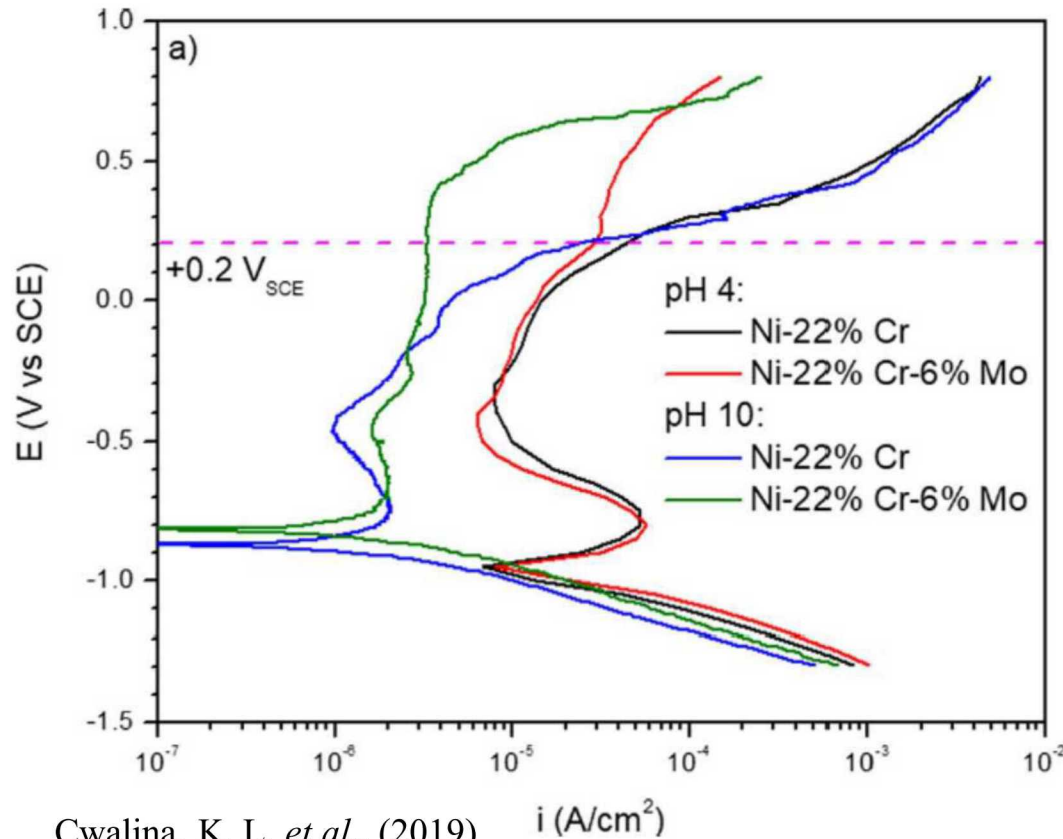
PRESENTED BY

Michael Melia (SNL)

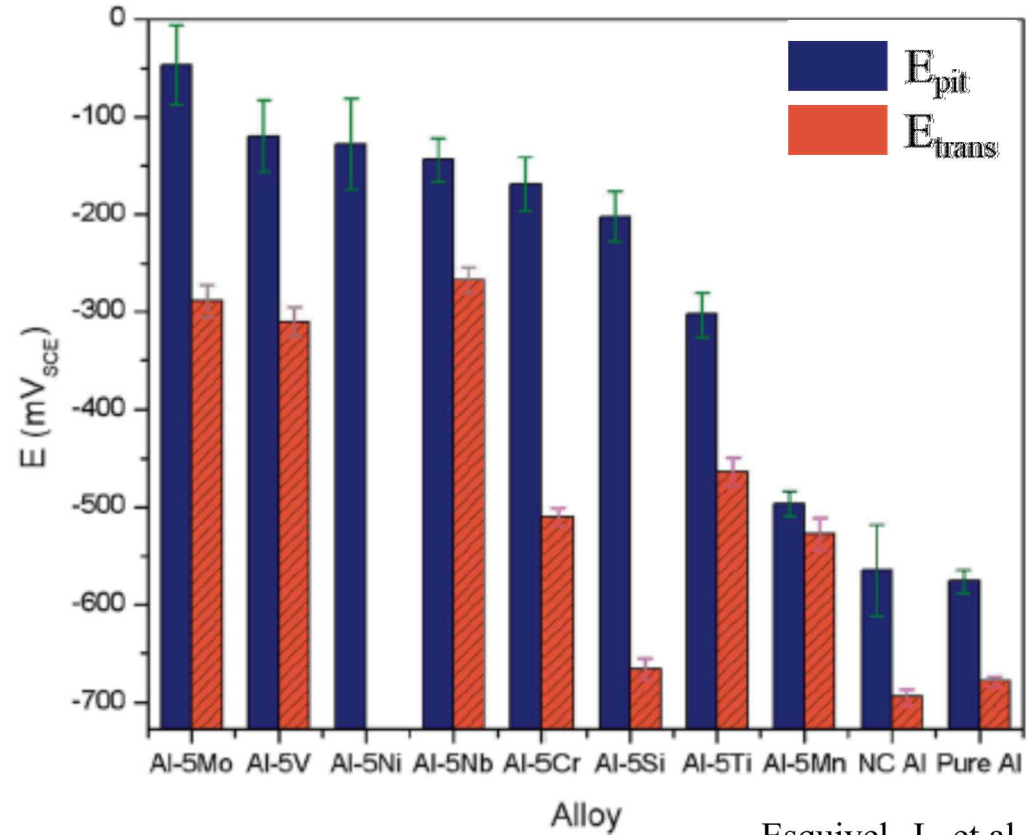
Co-authors: Jonathan Pegues, Morgan Jones, Brendan Nation, Nicolas Argibay, Andrew Kustas (SNL)

Kodi Summers and Dev Chidambaram (UN-Reno)

# Alloying impact on passivity



Cwalina, K. L. *et al.*, (2019)



Esquivel, J., *et al.* (2018)

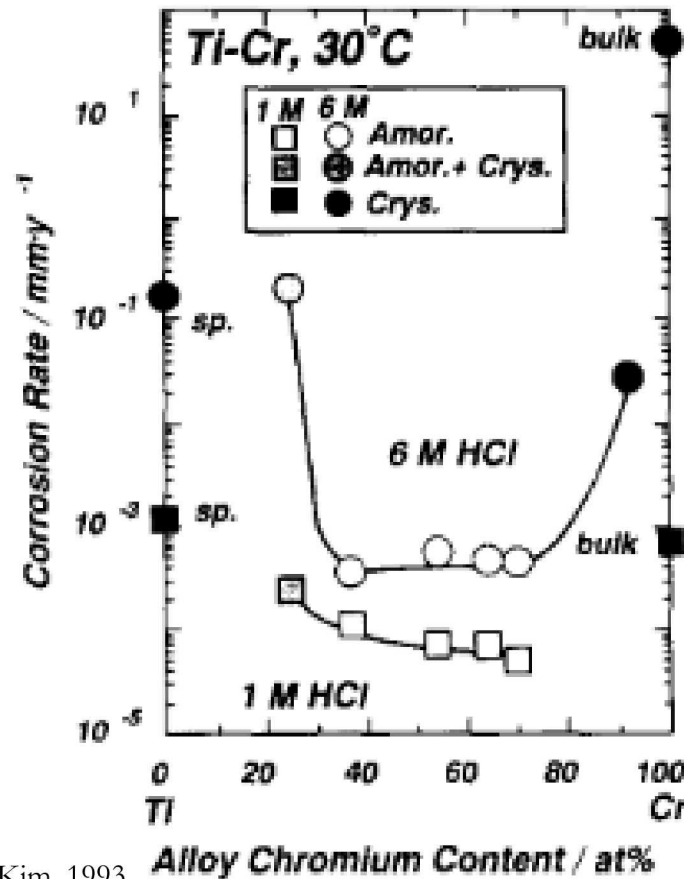
Passivity is heavily influenced by alloying elements, i.e. Cr/Mo additions to steel or Ni alloys.

Other passivity promoting elements (Ti, Mo, Nb, etc.) show beneficial effects when added to pure Al, assuming the elements are kept in solid solution.



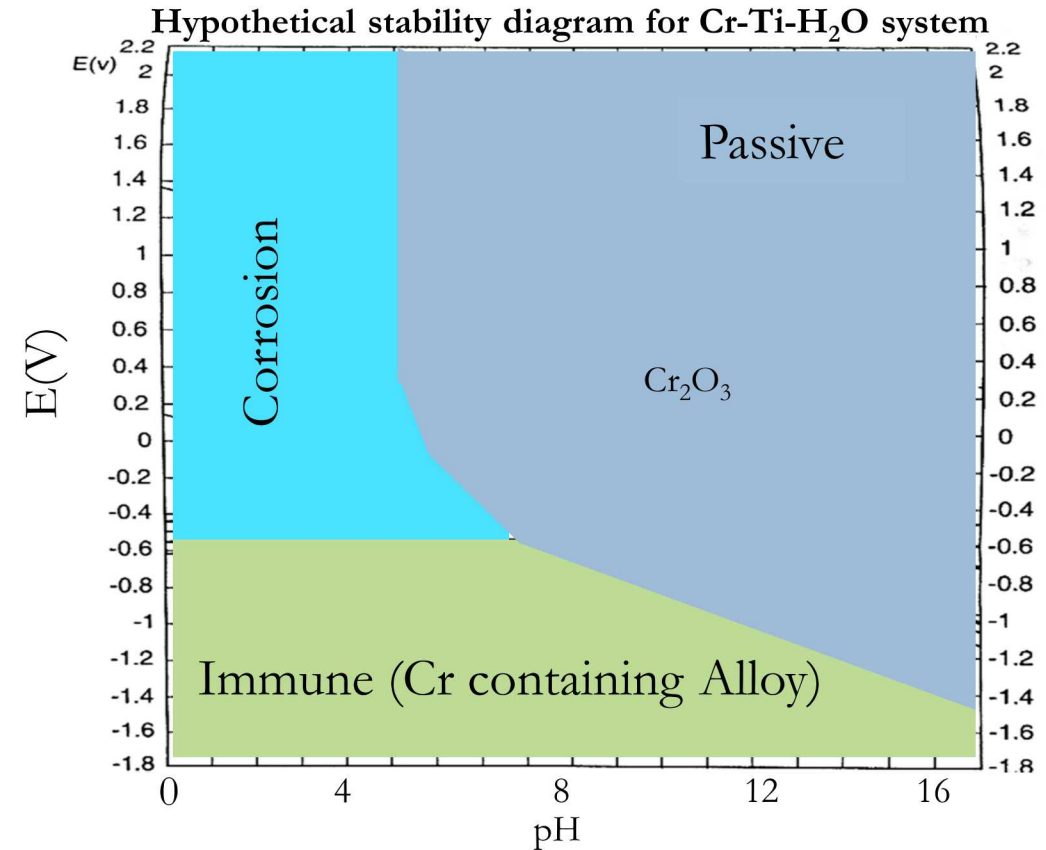
# Alloying impact on passivity – Cr-Ti system

## Concentrated HCl



J.H. Kim, 1993

The combination of Ti and Cr has been shown to have synergistic effects for corrosion resistance/passivity across a range of aggressive environments (concentrated acids).

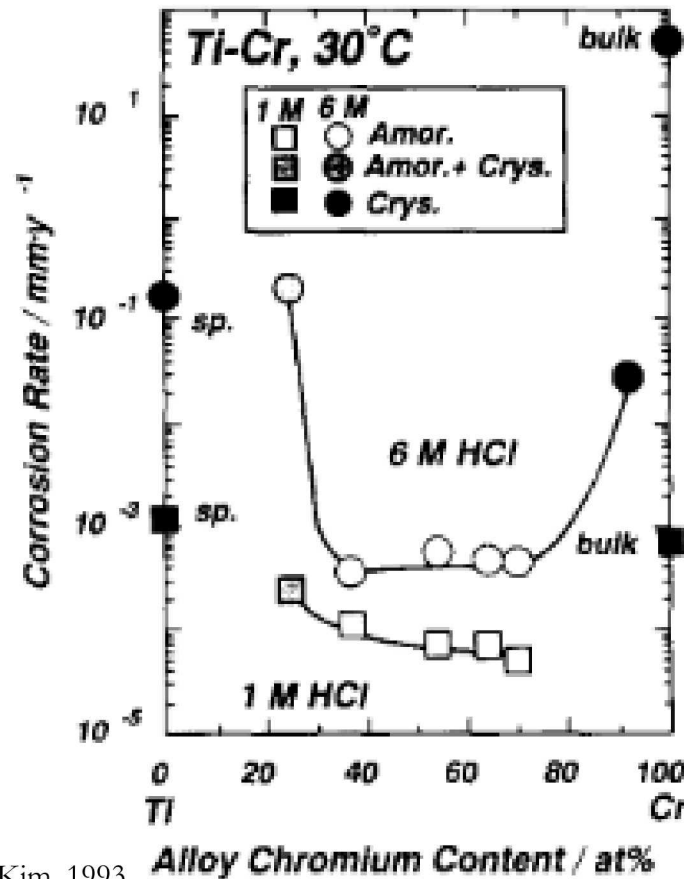


The underlying passivating mechanism is expected, for these Ti-Cr systems, to be caused by the formation of a homogeneous double oxy-hydroxide of Cr<sup>3+</sup> and Ti<sup>4+</sup>, stable across a larger pH and potential range than typically experienced by pure Ti or Cr.

M. Mehmood, 1999

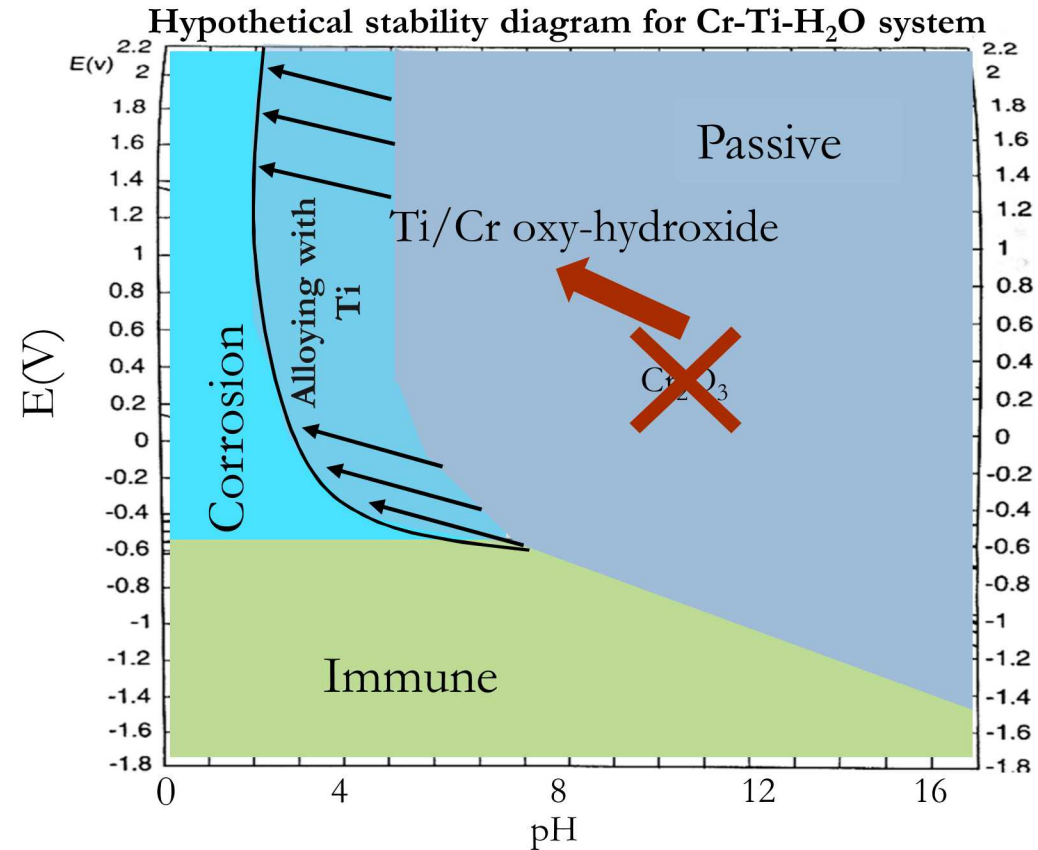
# Alloying impact on passivity – Cr-Ti system

## Concentrated HCl



J.H. Kim, 1993

The combination of Ti and Cr has been shown to have synergistic effects for corrosion resistance/passivity across a range of aggressive environments (concentrated acids).



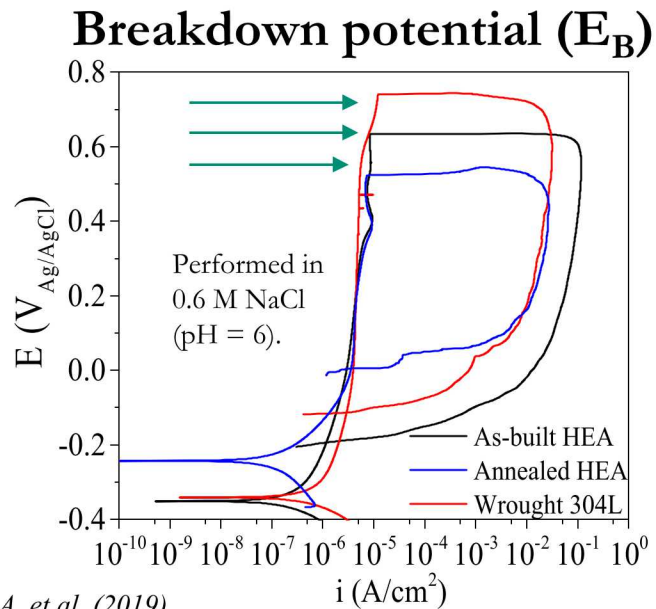
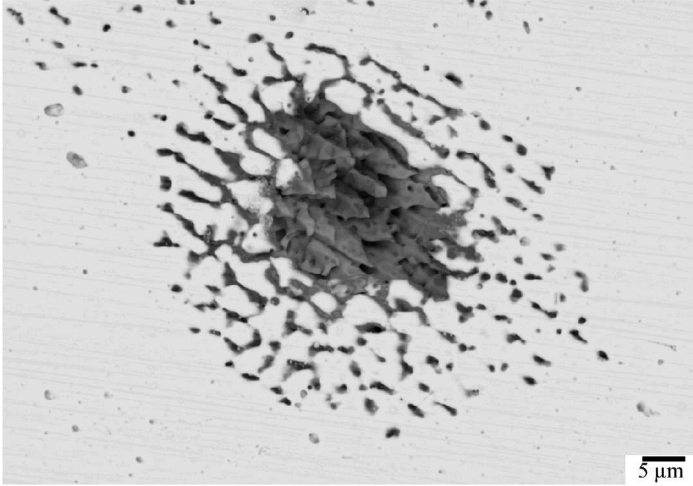
The underlying passivating mechanism is expected, for these Ti-Cr systems, to be caused by the formation of a homogeneous double oxy-hydroxide of Cr<sup>3+</sup> and Ti<sup>4+</sup>, stable across a larger pH and potential range than typically experienced by pure Ti or Cr.

M. Mehmood, 1999



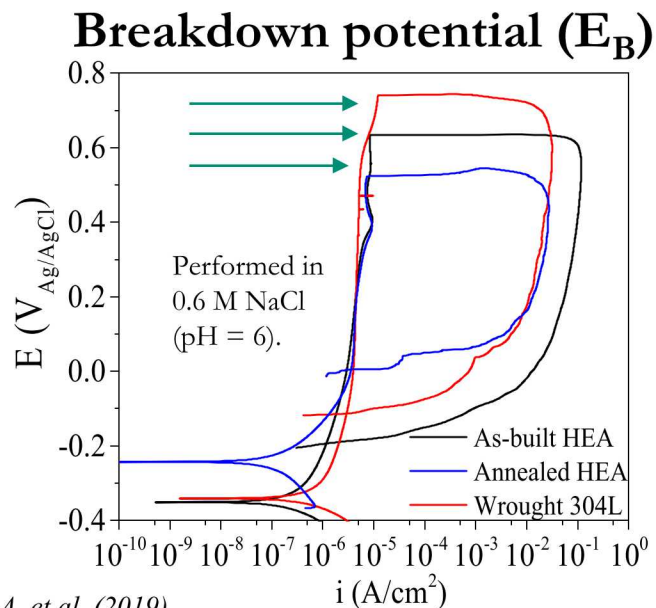
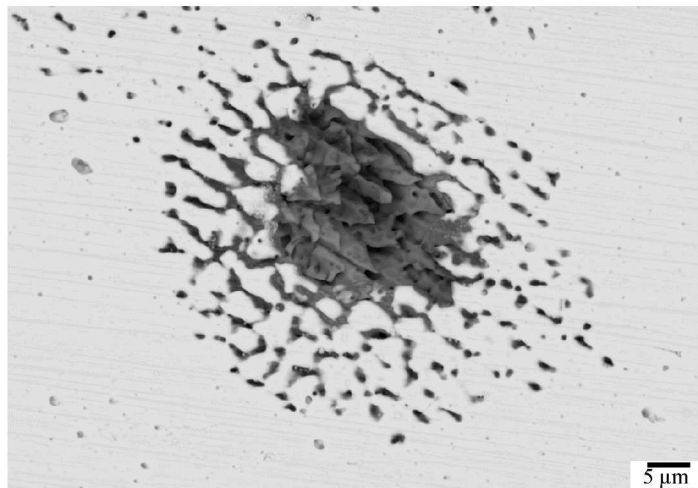
# Corrosion of high entropy alloys (HEAs)

## Pit morphology of CoCrFeMnNi HEA

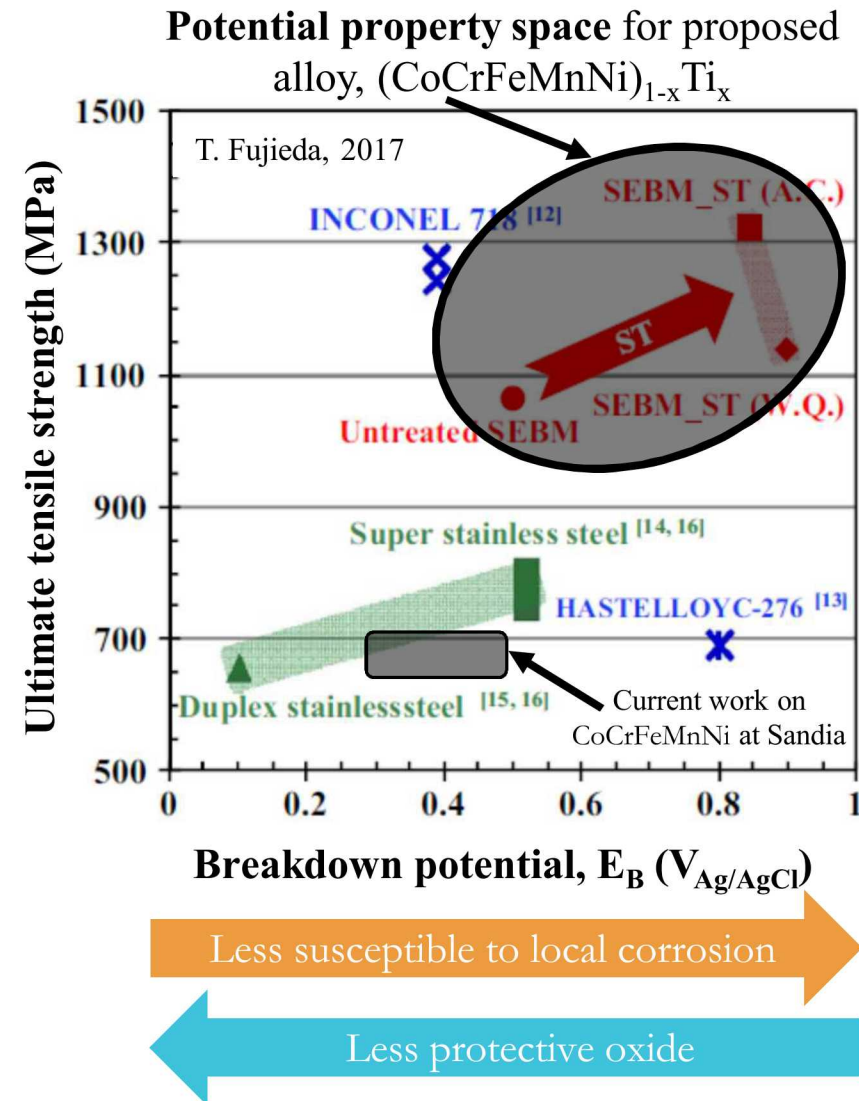


# Corrosion of high entropy alloys (HEAs)

## Pit morphology of CoCrFeMnNi HEA



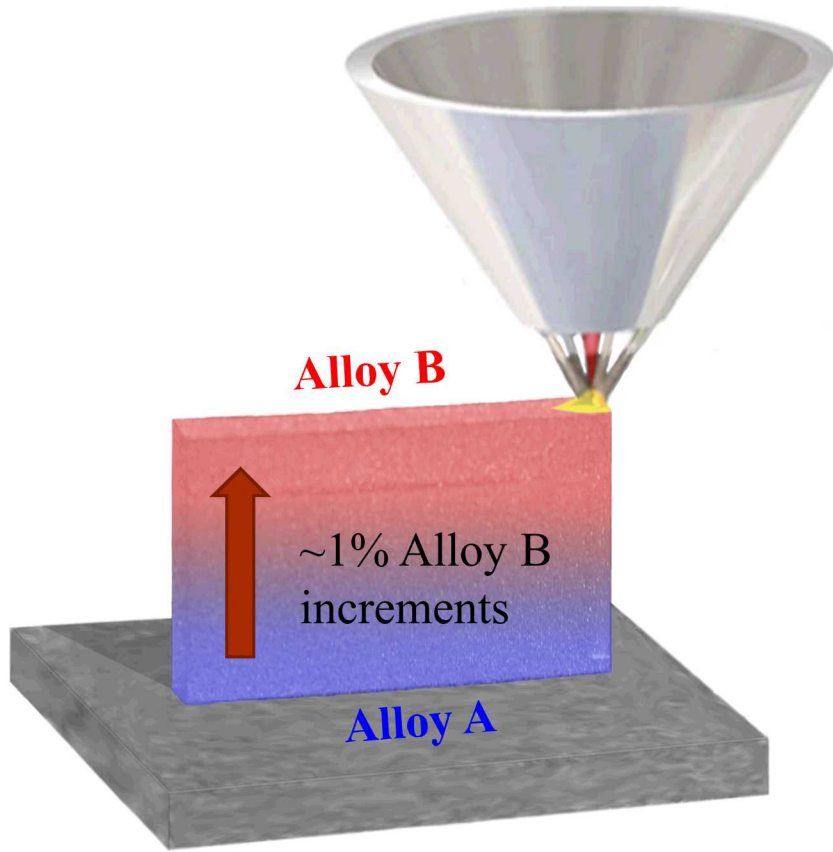
Melia, M. A. et al. (2019)



# High-throughput Alloy Screening by AM



Laser Beam – Directed Energy Deposition (LB-DED)



Samples were printed with an in-house LB-DED system with potential for mixing 5 powders.

A pre-mixed equiatomic CoCrFeMnNi alloy powder and commercial purity (CP) Ti powder were used.

Material characterization:

- Site specific x-ray diffraction (XRD).
- Energy dispersive spectroscopy (EDS) and Electron backscattered diffraction (EBSD).

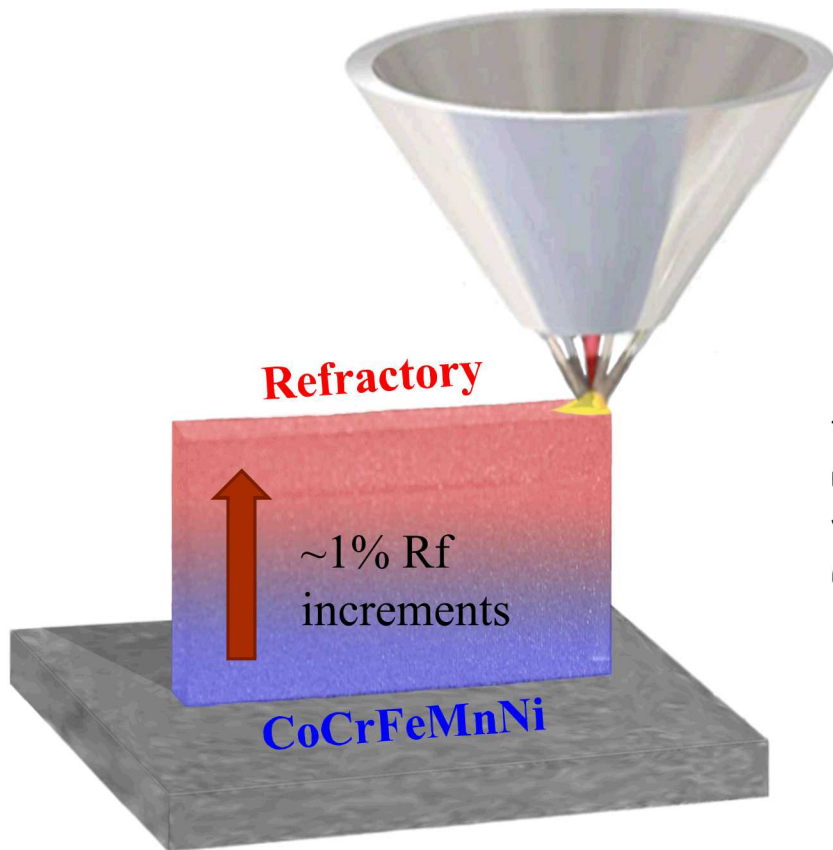
Electrochemical measurements:

- Capillary cell polarization measurements.
- Repassivation kinetics by scratch testing.

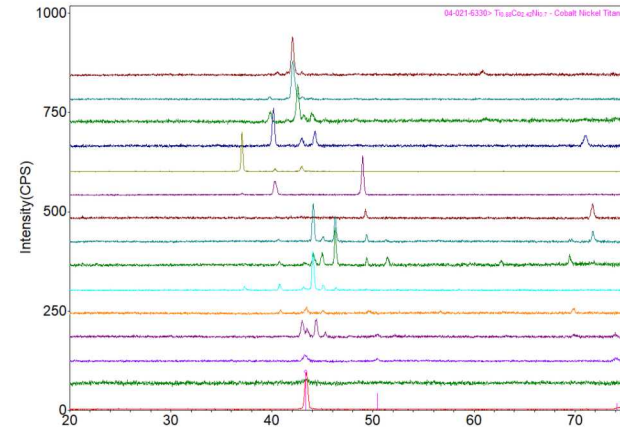


# High-throughput Alloy Screening

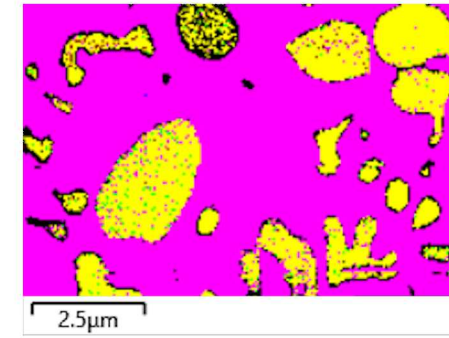
Laser Beam – Directed Energy Deposition (LB-DED)



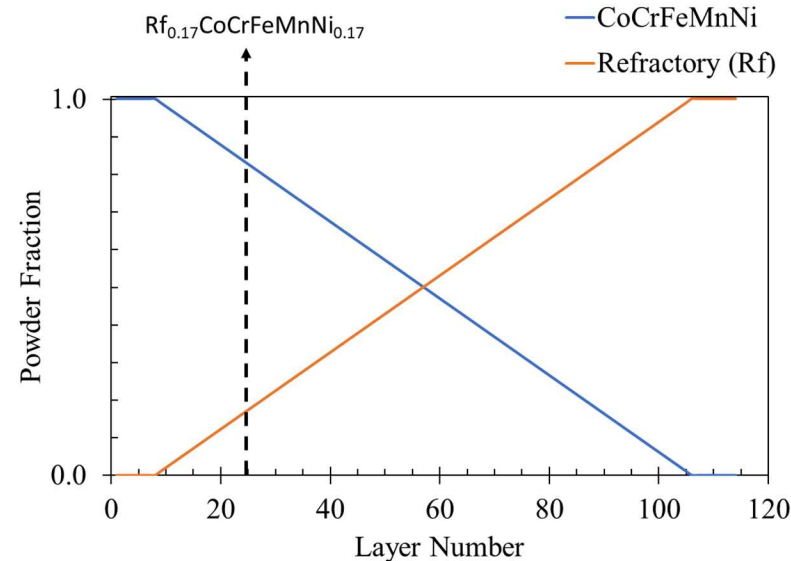
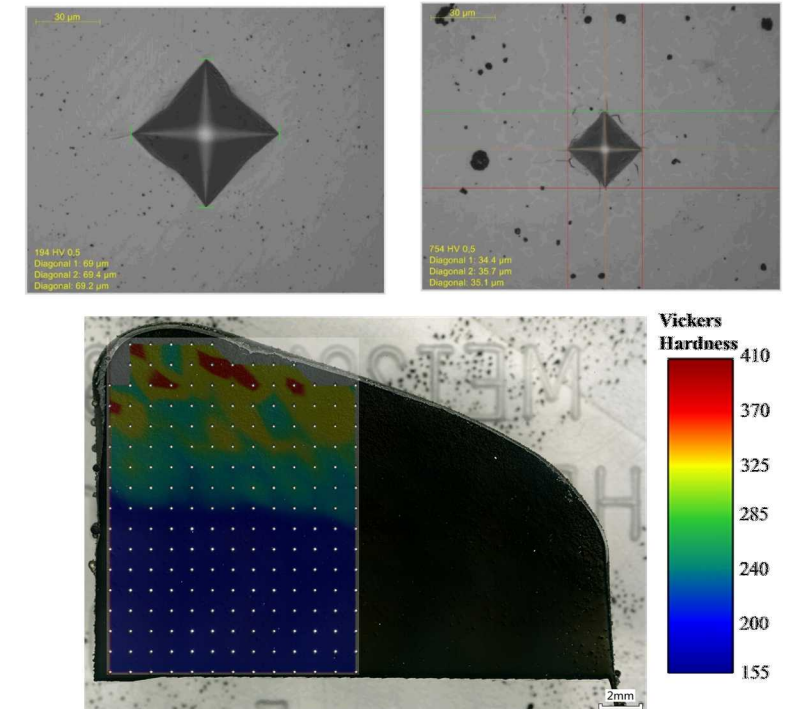
Site specific XRD



SEM – EDS/EBSD



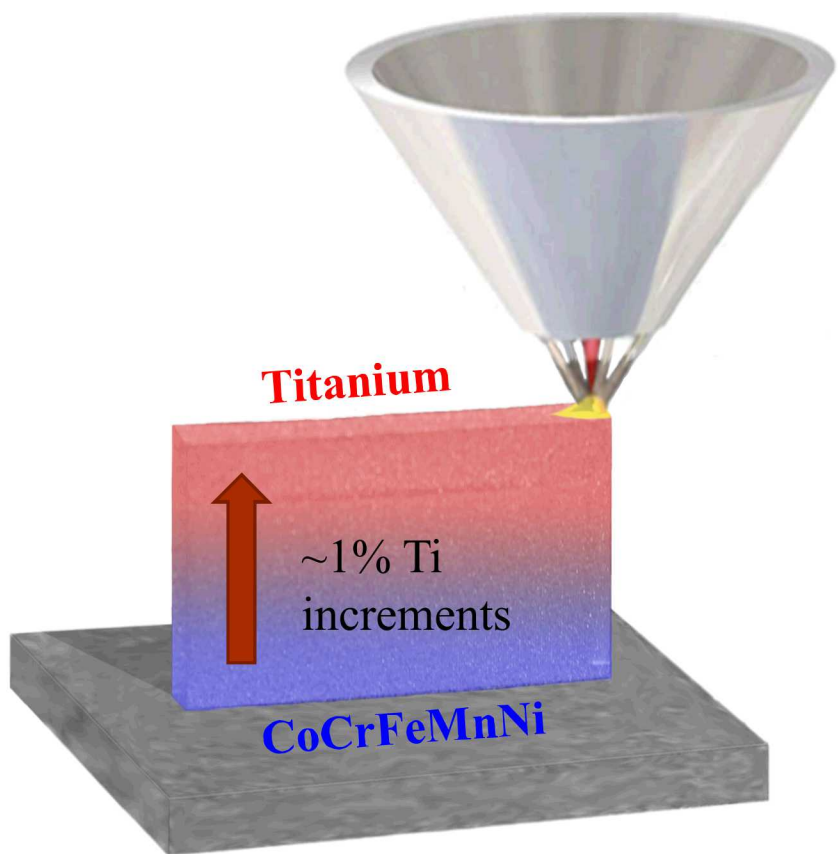
Vickers Hardness mapping



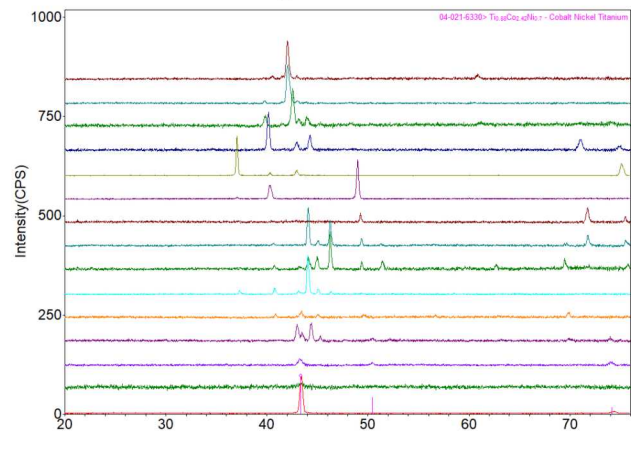
$$Rf_x[CoCrFeMnNi]_{(1-x)/5}$$

# High-throughput Alloy Screening – $\text{Ti}_x[\text{CoCrFeMnNi}]_{(1-x)/5}$

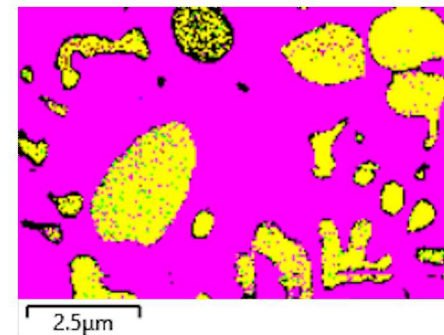
Laser Beam – Directed Energy Deposition (LB-DED)



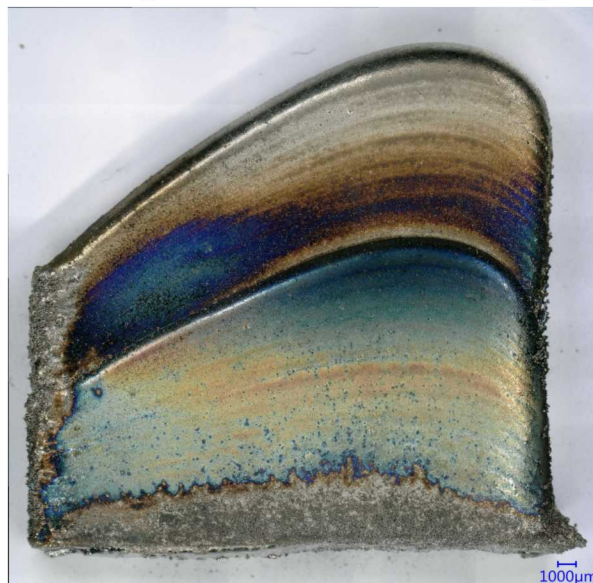
Site specific XRD



SEM – EDS/EBSD



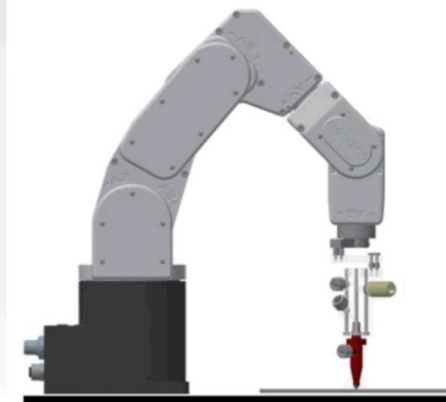
Actual printed HEA/Ti sample



Repassivation kinetics (scratch testing)



Capillary cell electrochemistry

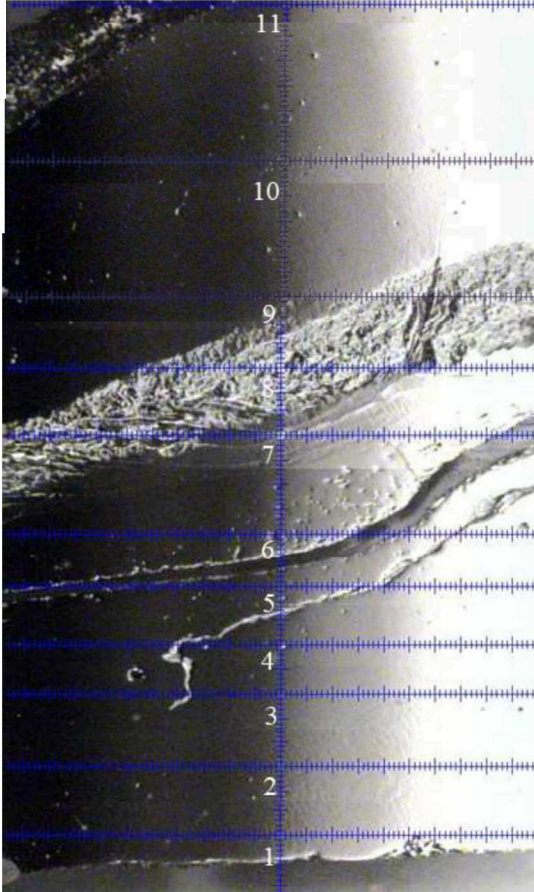


*Summers, K. & Chidambaram, D. (2020)*

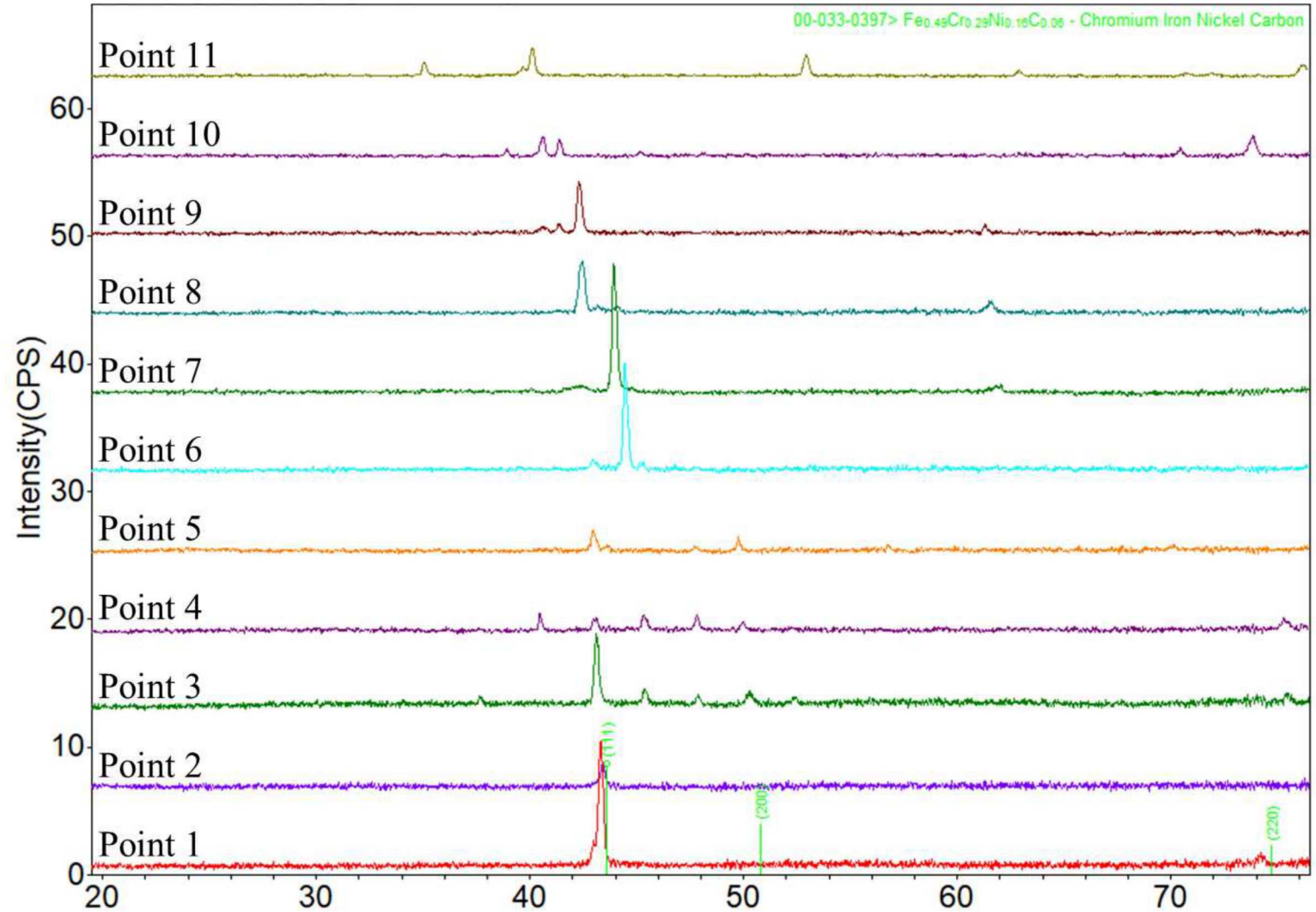


# $\text{Ti}_x[\text{CoCrFeMnNi}]_{(1-x)/5}$ – graded composition phase analysis

Pure Ti



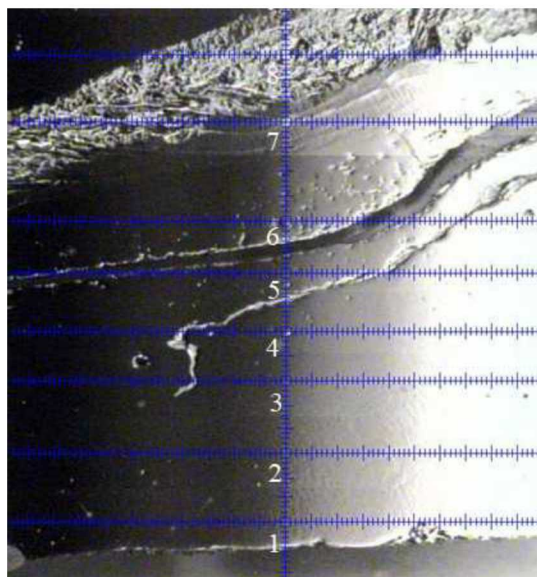
Pure CoCrFeMnNi alloy





# $\text{Ti}_x[\text{CoCrFeMnNi}]_{(1-x)/5}$ – graded composition phase analysis

Pure Ti



Pure CoCrFeMnNi alloy

Hexagonal Ti + Ti rich BCC intermetallic

BCC Ti + cubic  $\text{Ti}_2\text{Ni}$  – Ni component is like a combination of other transition metals.

BCC NiTi + BCC Ti + cubic  $\text{Ti}_2\text{Ni}$

BCC NiTi + hexagonal  $\text{TiFe}_2$

BCC  $\text{Ti}_{0.16}\text{Cr}_{0.26}\text{Fe}_{.58}$  + hexagonal TiCrMn (similar to  $\text{TiFe}_2$ )

BCC  $\text{Ti}_{0.16}\text{Cr}_{0.26}\text{Fe}_{.58}$  + hexagonal  $\text{TiFe}_2$

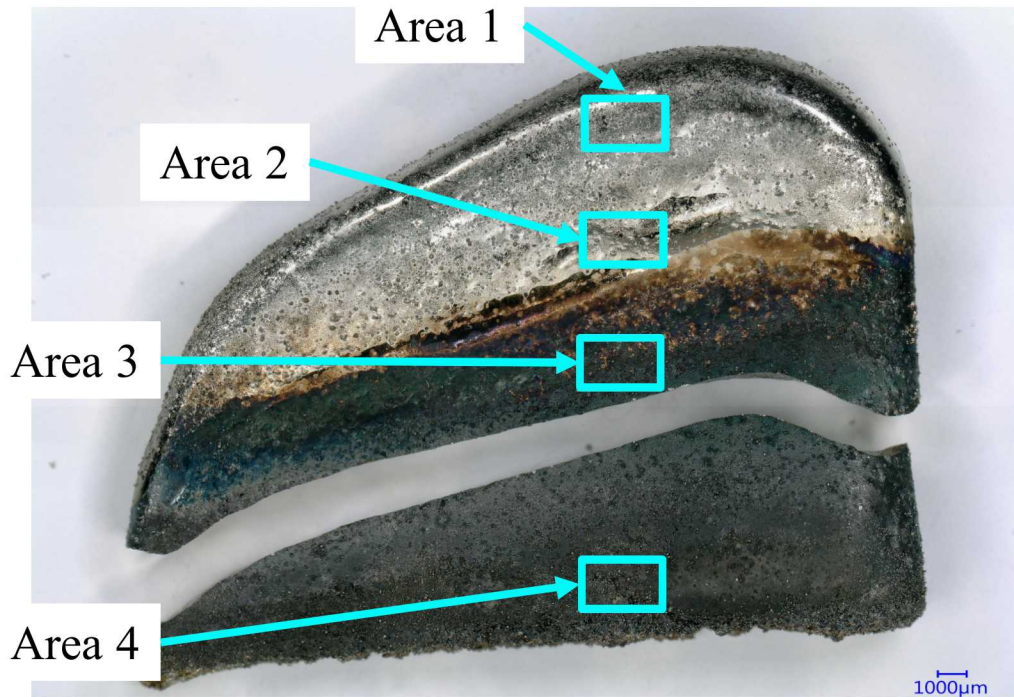
FCC steel + BCC  $\text{Ti}_{0.16}\text{Cr}_{0.26}\text{Fe}_{.58}$  + hexagonal  $\text{TiFe}_2$

FCC steel



# $\text{Ti}_x[\text{CoCrFeMnNi}]_{(1-x)/5}$ – graded composition phase analysis

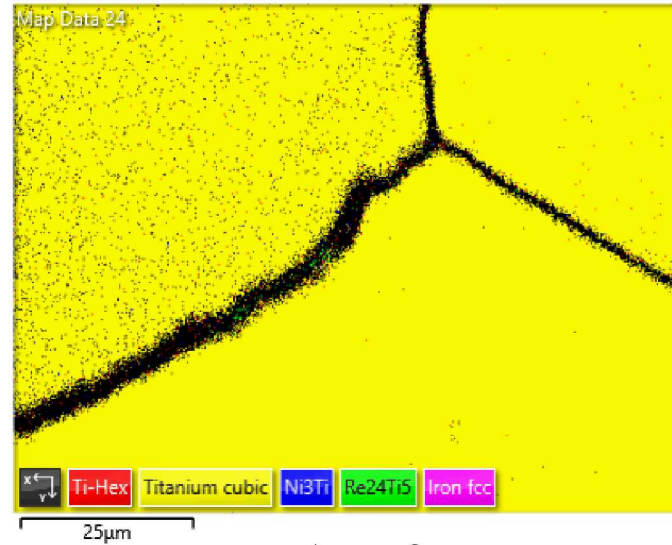
Ti BCC 
  Ti Hex 
   $\text{Ni}_3\text{Ti}$   
  $\text{Ti}_{24}\text{Ti}_5$ 
 Fe FCC



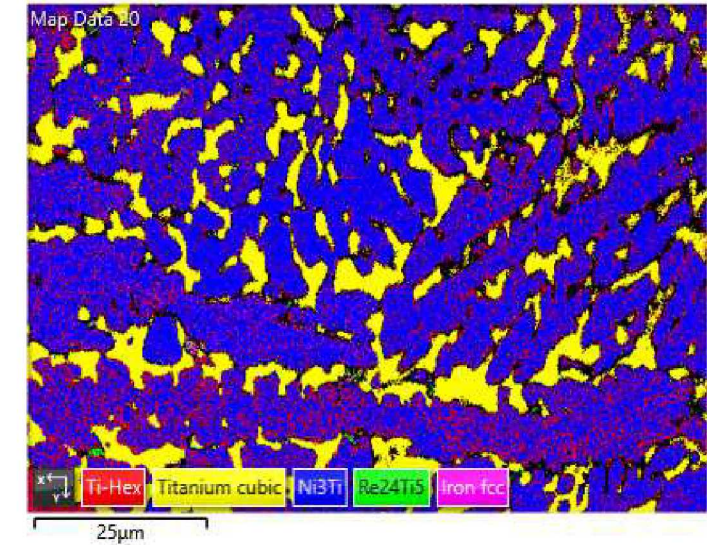
Using EBSD alone does not allow for accurate phase identification, analysis of material with XRD informed phase ID appears to be more accurate.

XRD helps distinguish between Ti rich and transition metal rich intermetallic phases.

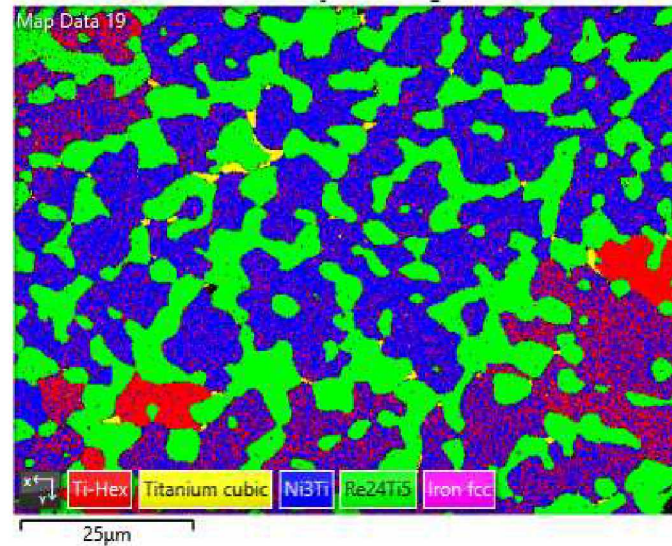
Area 1



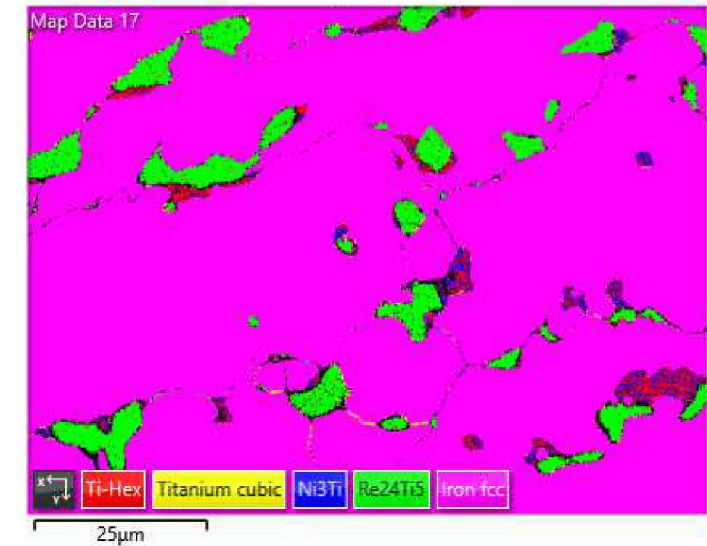
Area 2



Area 3

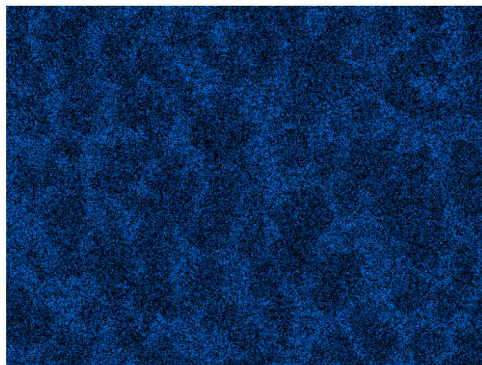
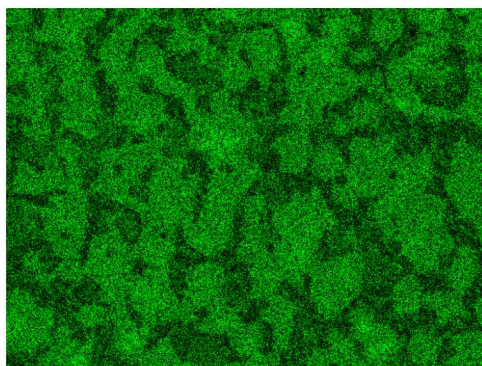
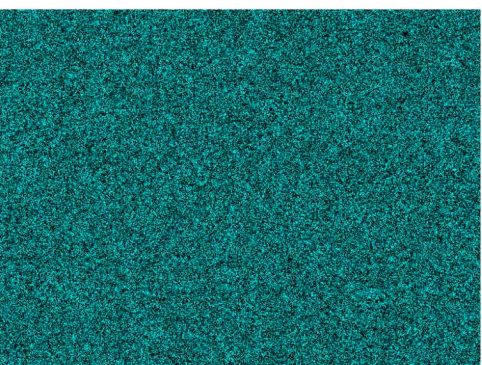
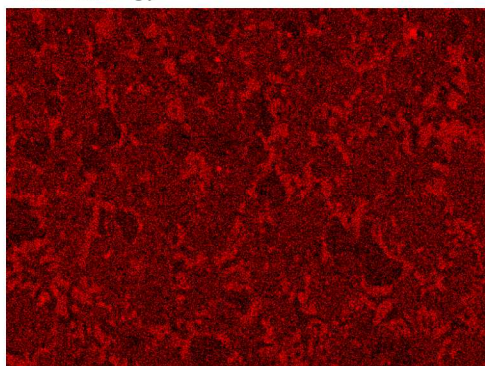
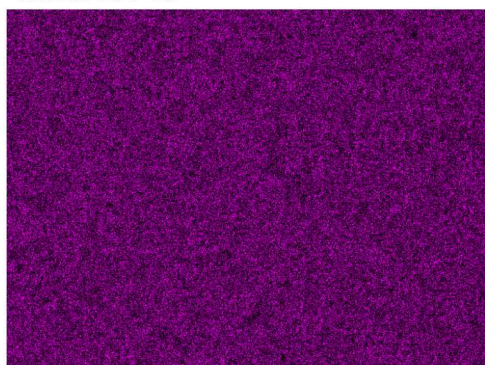


Area 4

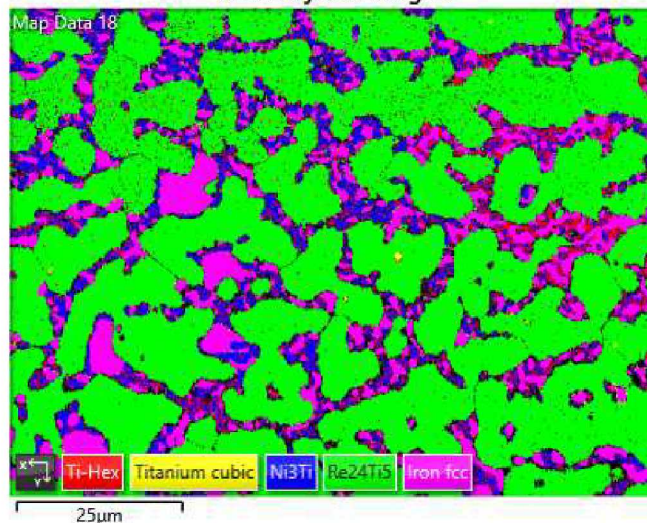




# $\text{Ti}_x[\text{CoCrFeMnNi}]_{(1-x)/5}$ – Equi-atomic composition

Ni K- $\alpha$ Cr K- $\alpha$ Co K- $\alpha$ Ti K- $\alpha$ Mn K- $\alpha$ 

EBSD Layered Image 4

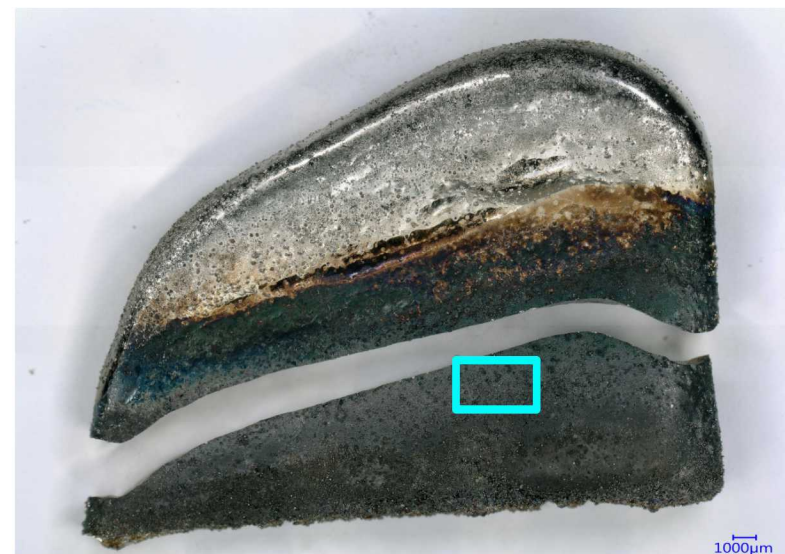


Expected phases from  
XRD/EBSD/EDS

■ BCC  $\text{Ti}_{0.16}\text{Cr}_{0.26}\text{Fe}_{.58}$

■ Hexagonal  $\text{TiFe}_2$

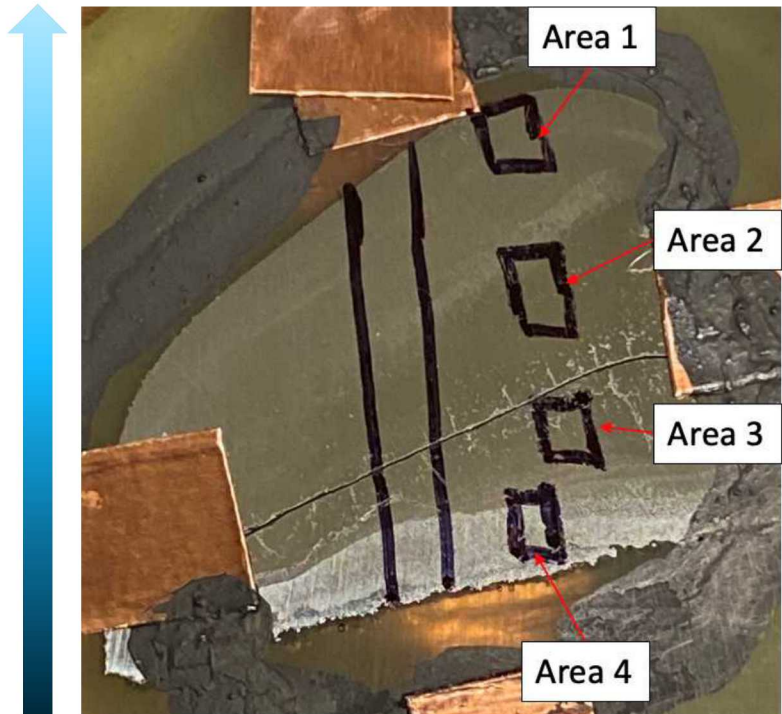
■ FCC - steel





# Electrochemical behavior – Capillary cell measurements

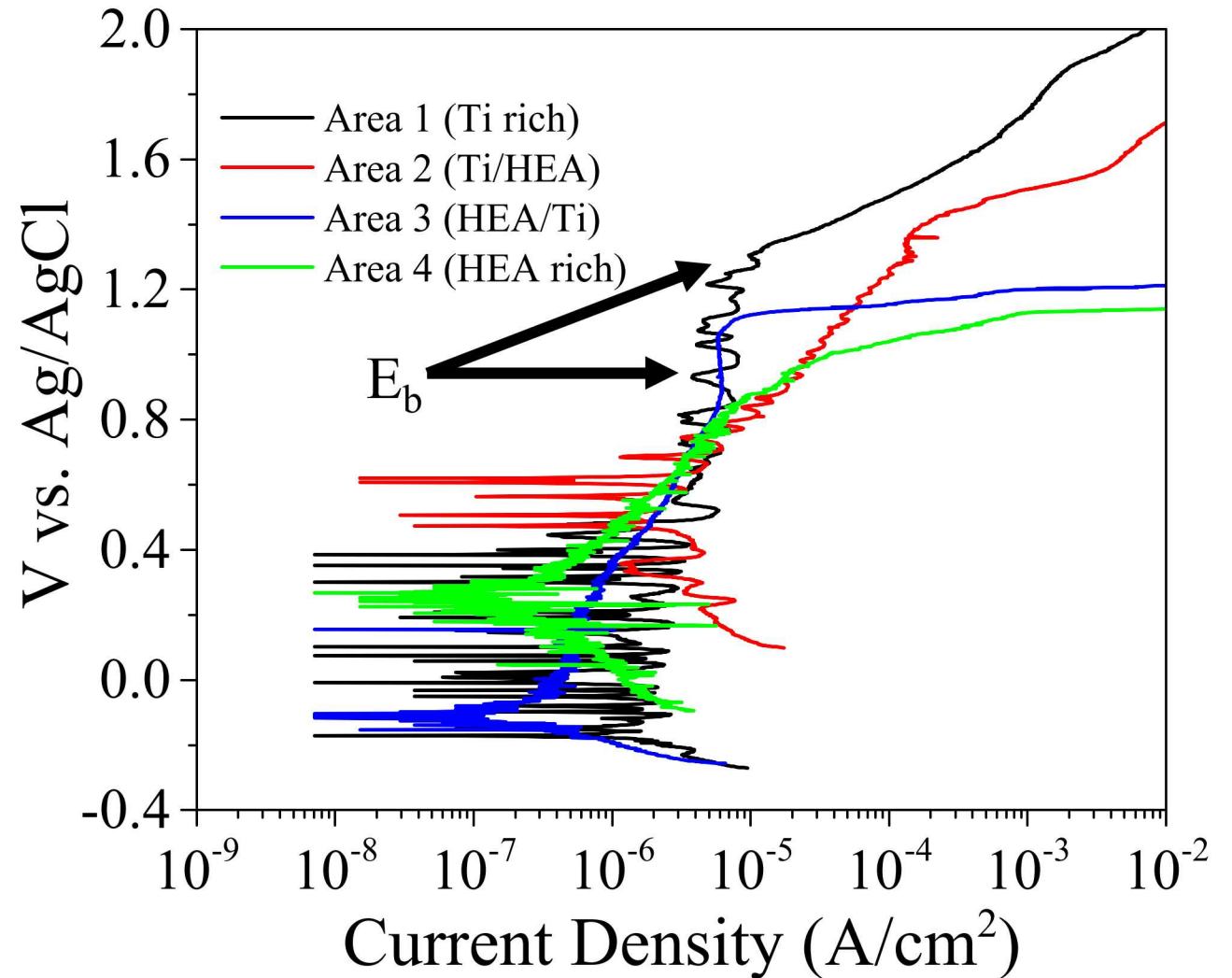
Pure Ti



Pure CoCrFeMnNi alloy

## Parameters

WE	Ti-graded Cantor
CE	Pt wire
RE	3M Ag AgCl
Electrolyte	0.599 M NaCl
Scan Range	-0.05V vs. E <sub>oc</sub> to 2.0V vs. E <sub>ref</sub>
Scan rate	1 mV/s
Flow rate	0.075 mL/min



# Electrochemical behavior – Capillary cell measurements

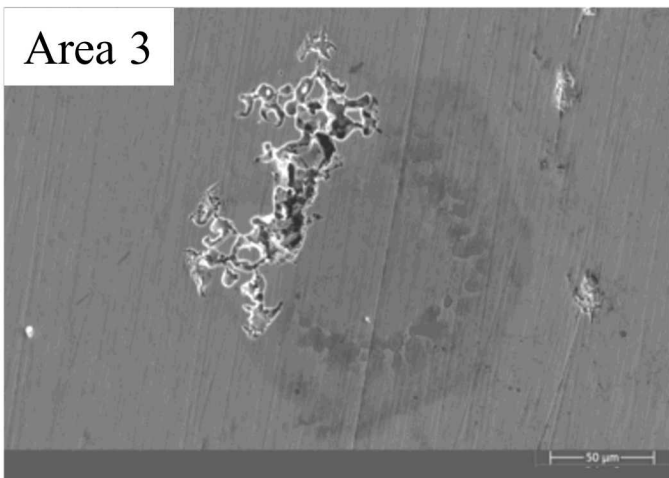
Area 1

*No image*

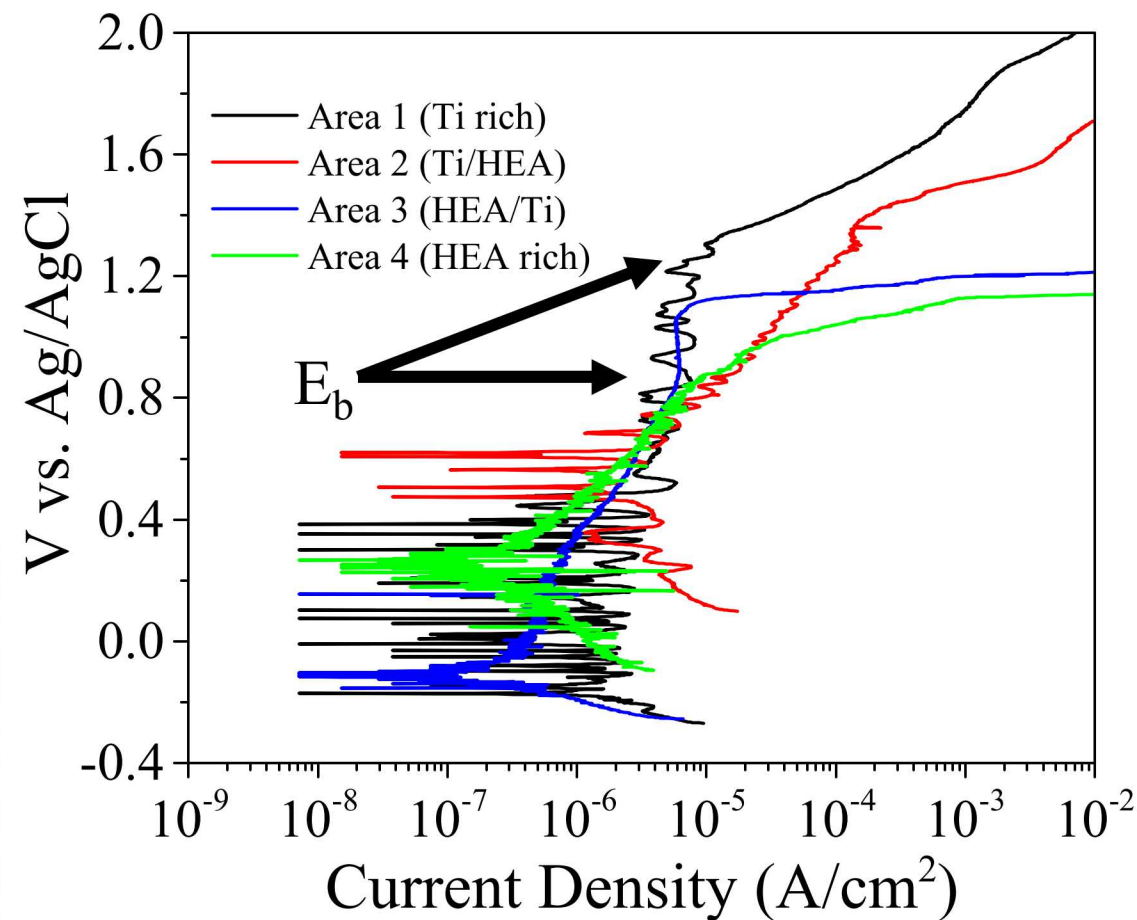
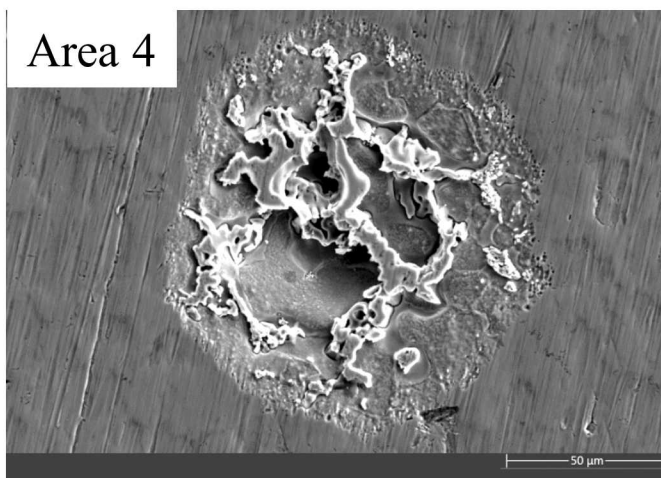
Area 2



Area 3



Area 4



**Area 1**  $E_b \sim 1.3$  V vs. Ag/AgCl

**Area 2**  $E_b \sim 1.3$

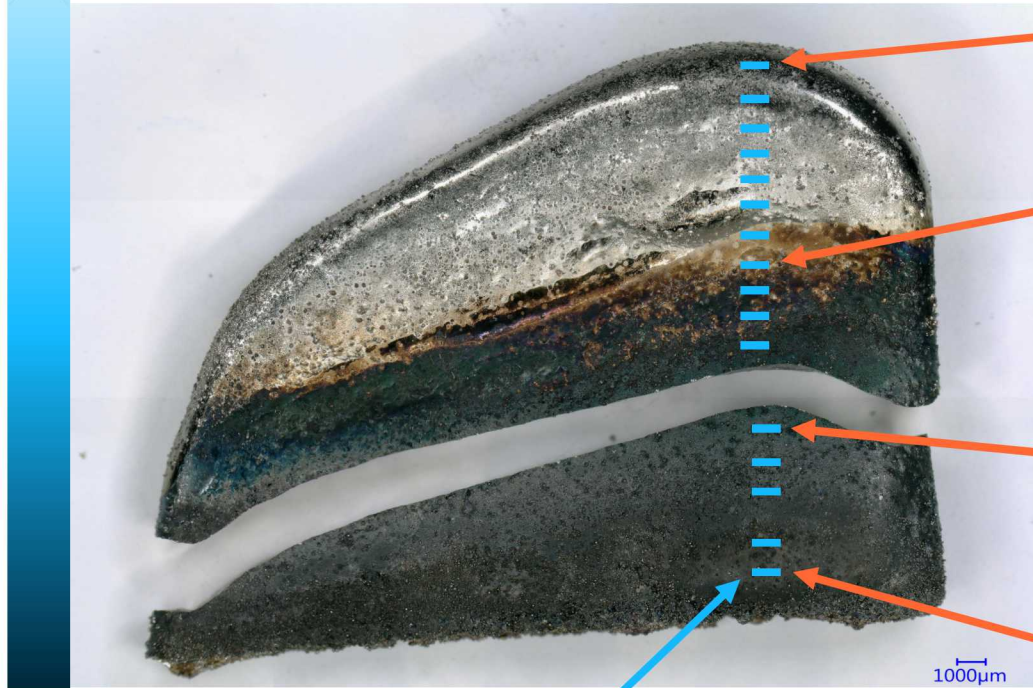
**Area 3**  $E_b \sim 1.1$

**Area 4**  $E_b \sim 0.9$



# Electrochemical behavior – Scratch testing for repassivation kinetics

Pure Ti



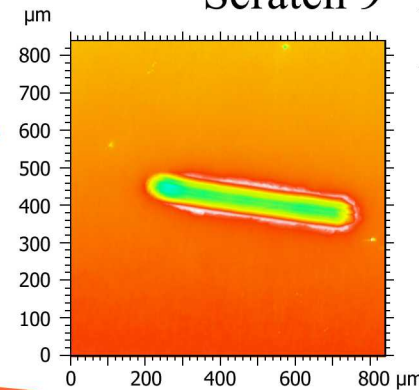
Pure CoCrFeMnNi alloy

First scratch after 5500 second anodic polarization (+ 200 mV vs. Ag/AgCl) in 1 M Na<sub>2</sub>SO<sub>4</sub> solution.

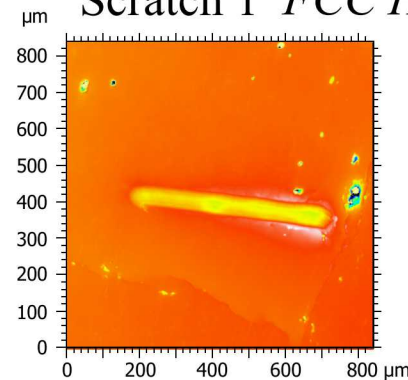
All performed with scratches at 20 N, 60 mm/min, 0.5 mm long.

SNL tribology lab – Brendan Nation, Morgan Jones, and Nic Argibay

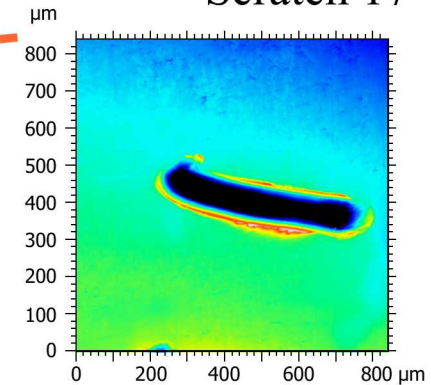
Scratch 9 *BCC NiTi + BCC Ti + cubic Ti<sub>2</sub>Ni*



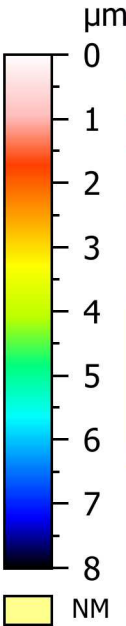
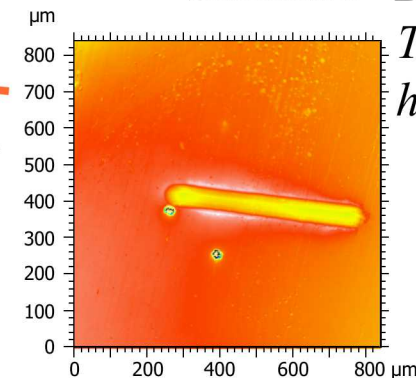
Scratch 1 *FCC HEA*



Scratch 17 *Hexagonal Ti + Ti rich BCC intermetallic*



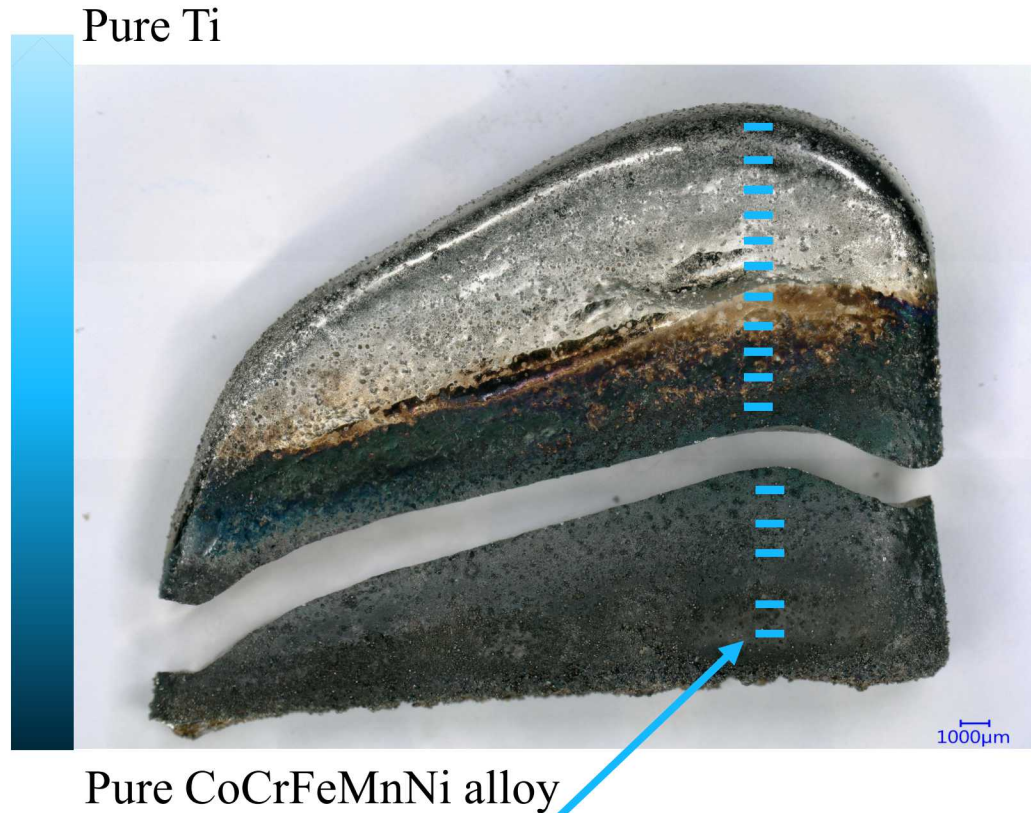
Scratch 5 *BCC Ti<sub>0.16</sub>Cr<sub>0.26</sub>Fe<sub>0.58</sub> + hexagonal TiFe<sub>2</sub>*



17 scratches by a spheroconical diamond scribe ( $r = 100 \mu\text{m}$ ) along the composition gradient.

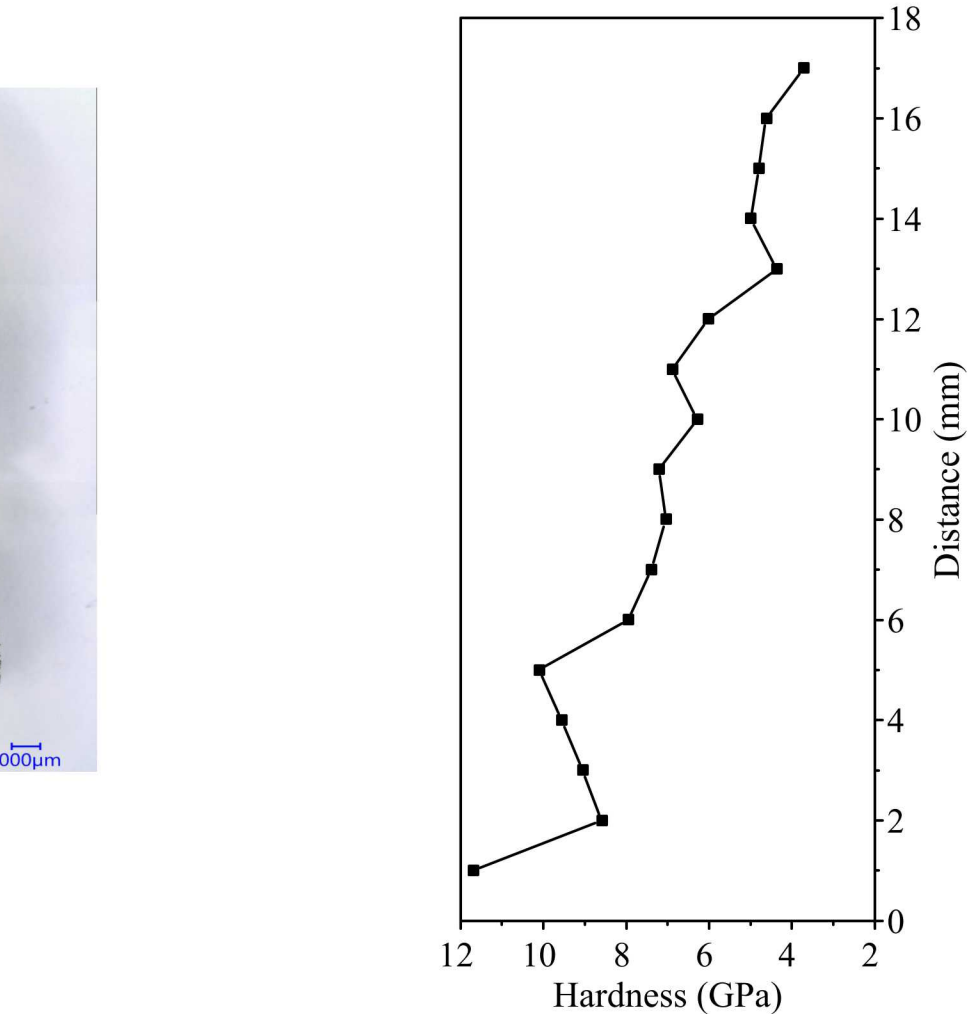


# Electrochemical behavior – Scratch testing for hardness measurement



First scratch after 5500 second anodic polarization (+ 200 mV vs. Ag/AgCl) in 1 M Na<sub>2</sub>SO<sub>4</sub> solution.

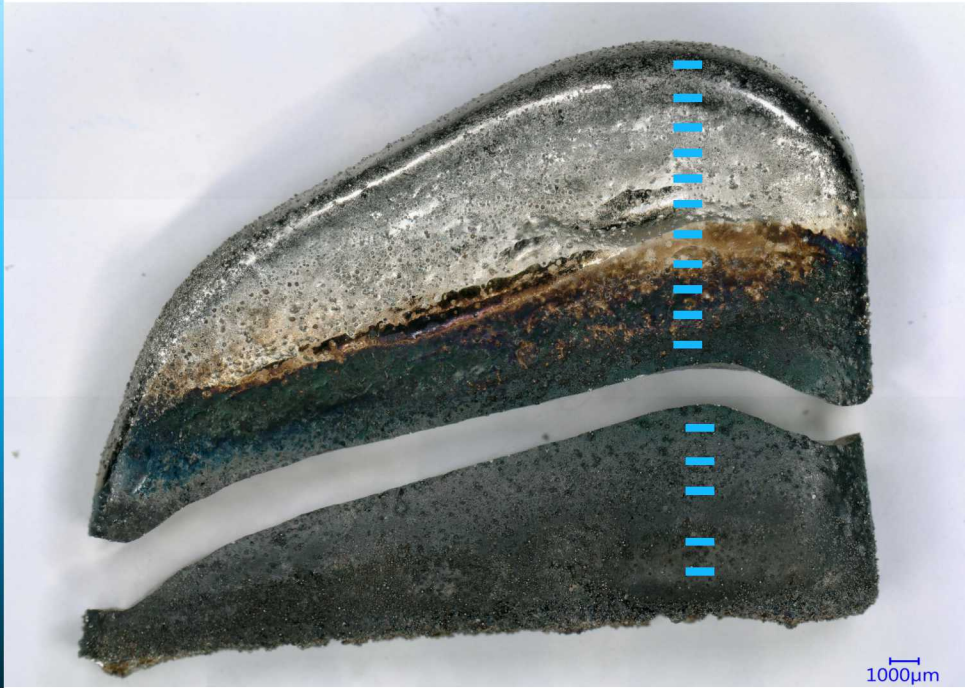
All performed with scratches at 20 N, 60 mm/min, 0.5 mm long.



*17 scratches by a spheroconical diamond scribe ( $r = 100 \mu\text{m}$ ) along the composition gradient.*

# Electrochemical behavior – Scratch testing for hardness measurement

Pure Ti

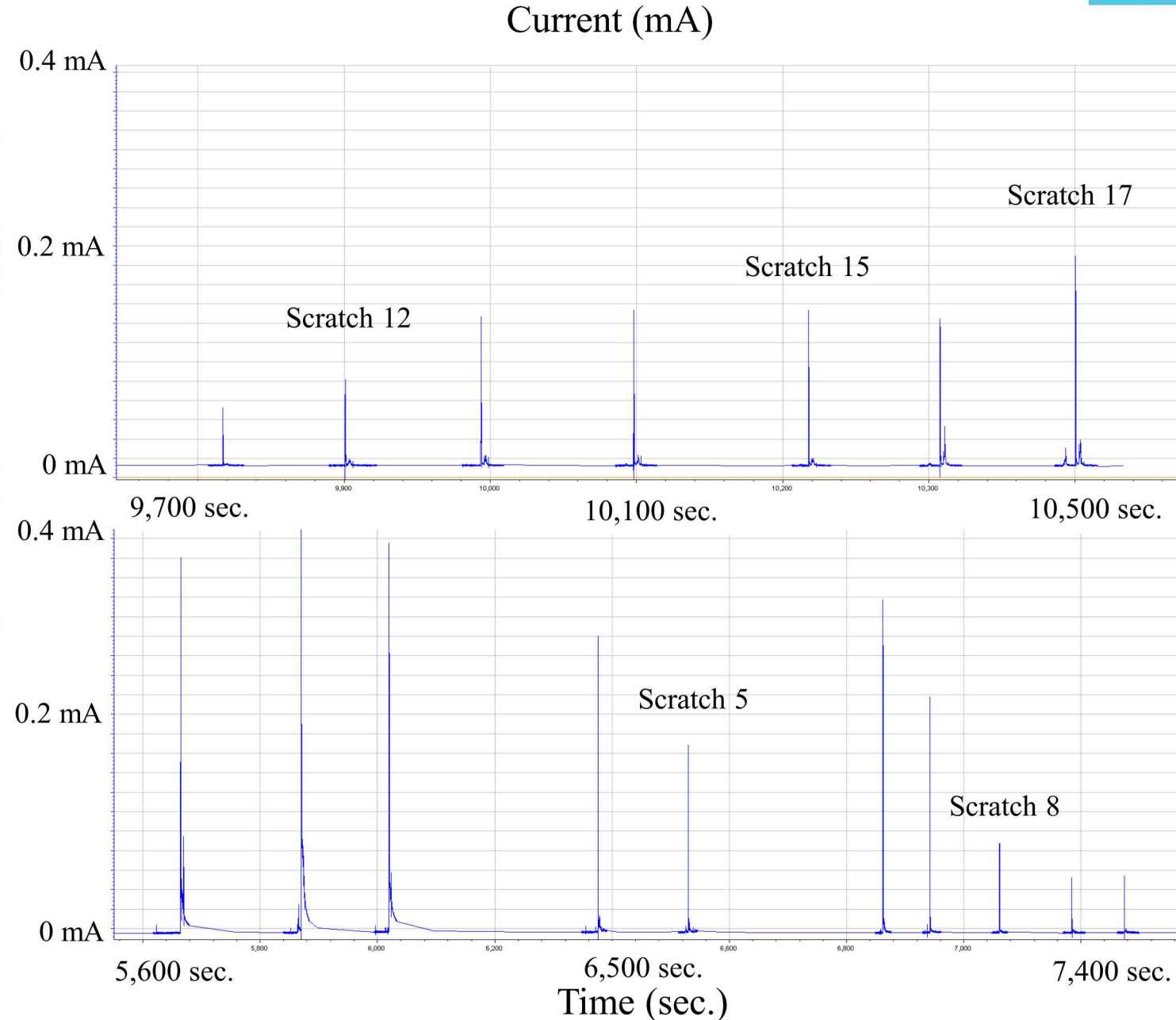


Pure CoCrFeMnNi alloy

Prior to the scribe touching the sample, the data acquisition rate was switched to 10,000 pt./sec.

Peak current is maximum for the CoCrFeMnNi rich side of the sample, minimum at the Ti rich/IMC rich region, and increased when moving closer to pure Ti.

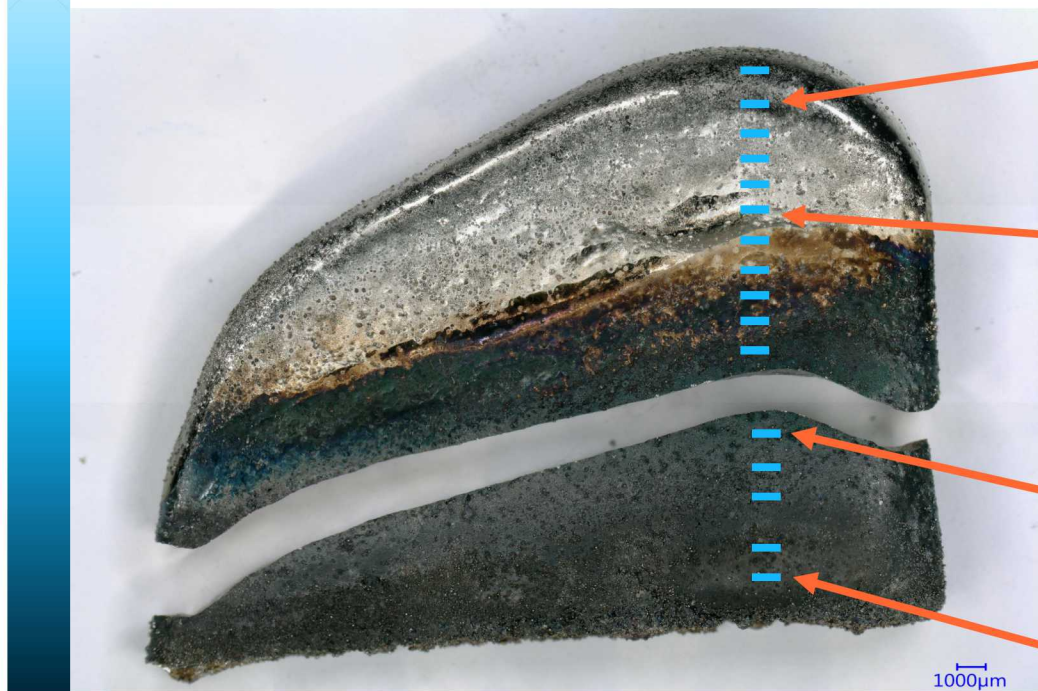
SNL tribology lab – Brendan Nation, Morgan Jones, and Nic Argibay





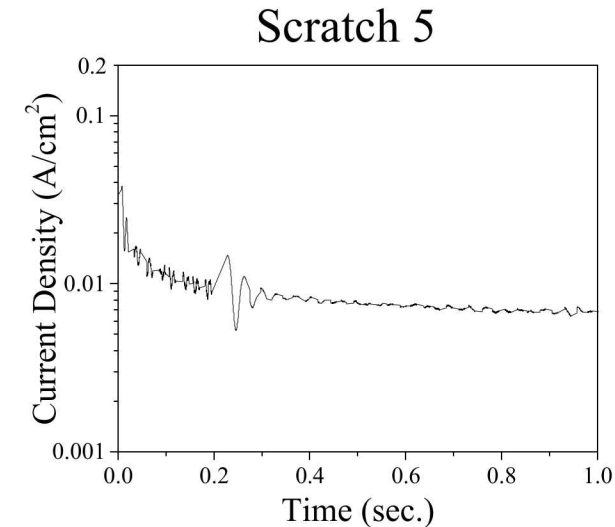
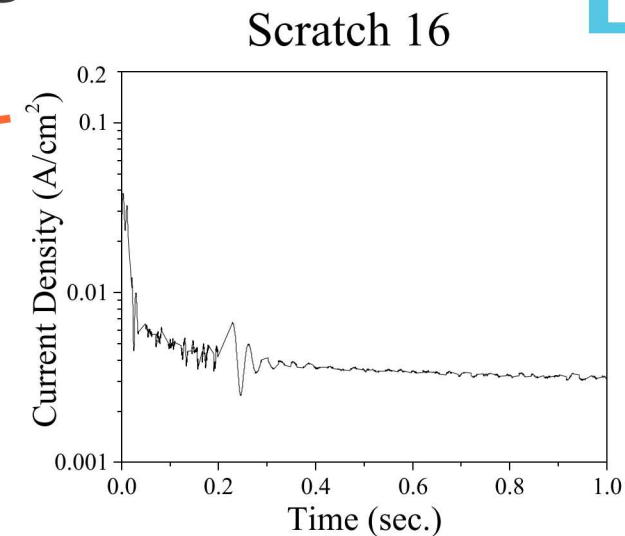
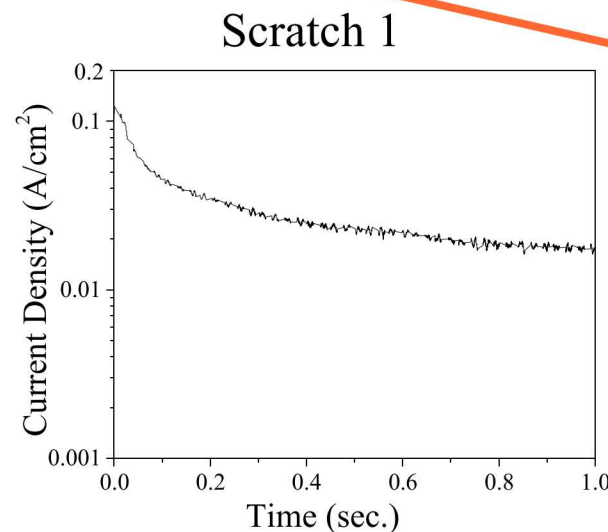
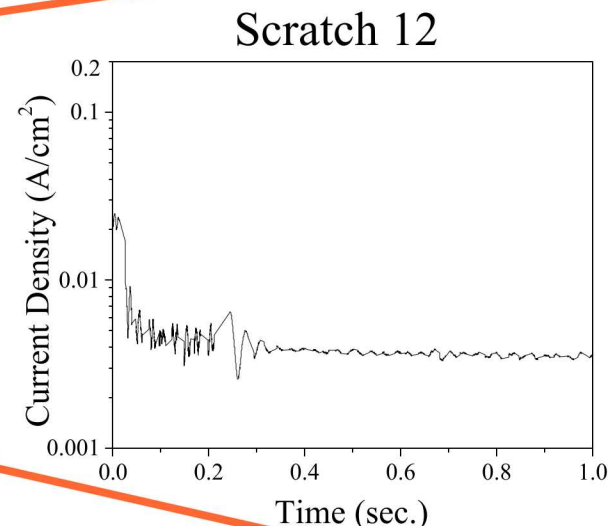
# Electrochemical behavior – Scratch testing for repassivation kinetics

Pure Ti



Pure CoCrFeMnNi alloy

There is a lot of potential for this rapid experimental approach, but fine tuning will be required in the acquisition and analysis.



# Conclusions

- The phase analysis portion of the high throughput alloy screening approach is able to accurately determine phase constitution across these compositionally graded alloys.
  - Doing site specific XRD first, prior to EDS/EBSD, helps with more site specific phase analysis.
- Electrochemical measurements using a capillary cell showed the breakdown potential of the Ti rich side of the sample to be larger than the CoCrFeMnNi rich side of the sample.
  - Mitigating noise in measurements and limiting crevice corrosion will be critical to generating consistent results.
- Scratch testing shows promise as an approach to determine mechanical and corrosion properties of the compositionally graded alloys.
  - The quicker repassivation (return to stable current) for the Ti rich region of the sample correlates well with the larger breakdown potential seen for capillary cell measurements.



# Acknowledgements

Laboratory Directed Research and Development (LDRD) Programs funded this work.

## *Many people to thank...*

Shaun R. Whetten, Scotty Bobbitt, Michael Heiden, Raymond Puckett, Christina Profazi, Celedonio Jaramillo, Dustin Coleman, Sara Dickens, Alex Hickman, Luis Jauregui, and Timothy Ruggles.

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## Slide 21

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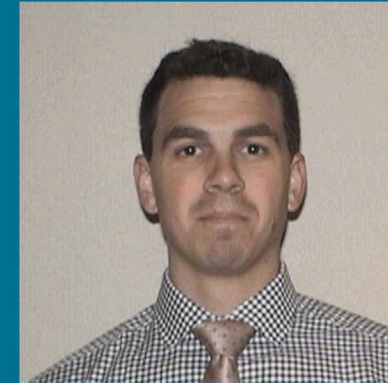
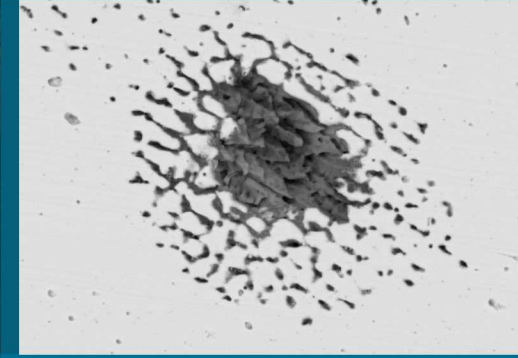
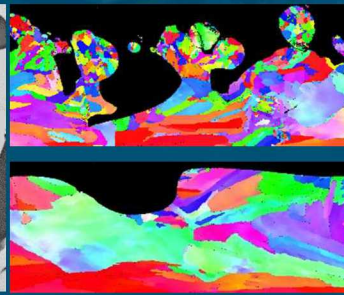
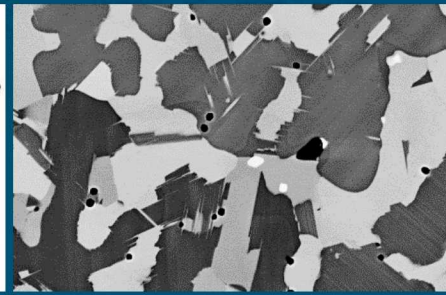
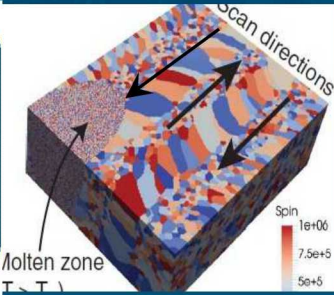
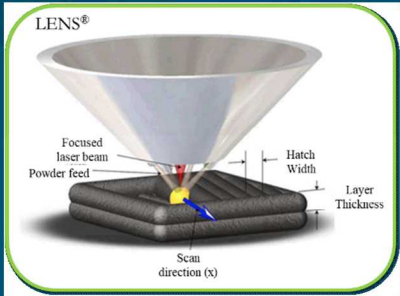
**SR13**

Probably don't need extra slides as you won't be able to present them...also ecs suggest putting your title slide again as your last slide

Schaller, Rebecca, 9/16/2020



# Additive Manufacturing to Compositionally Grade Metals for High Throughput Alloy Screening



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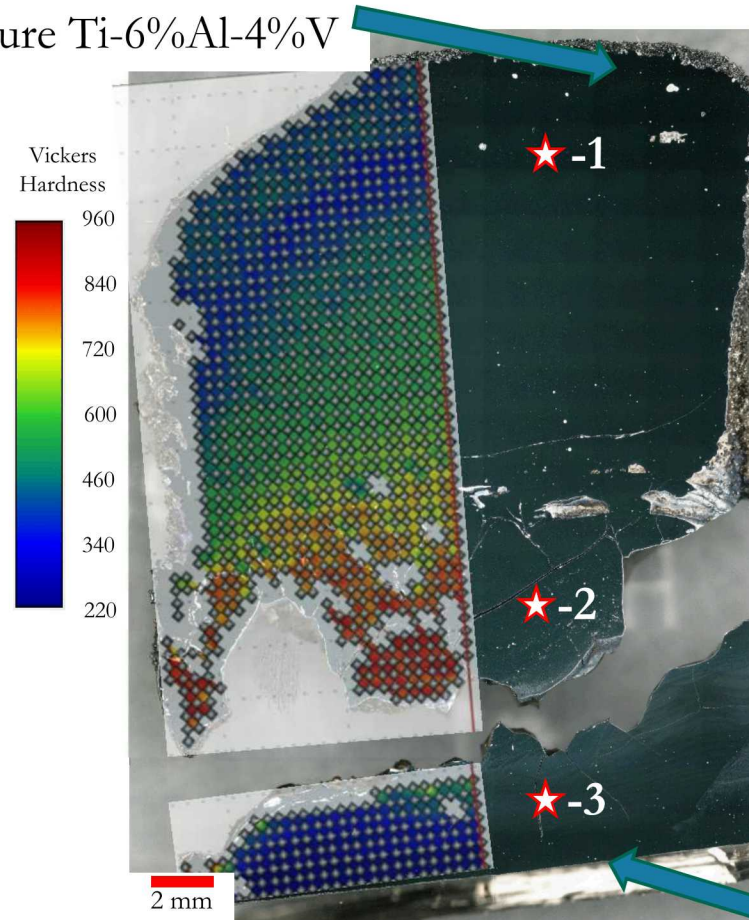
Extra slides



# Background: Current HEA studies at Sandia

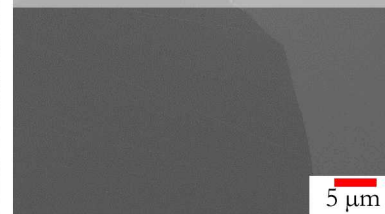
The most recent studies at Sandia used the LENS process to create a compositionally graded CoCrFeMnNi HEA with the Ti-6%Al-4%V alloy. These types of samples allow for rapid screening of properties such as microhardness (map in left of image).

Pure Ti-6%Al-4%V



Example electron micrographs

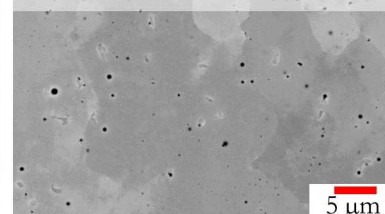
1)  $(\text{CoCrFeMnNi})_{10}\text{Ti}_{64}_{90}$



2)  $(\text{CoCrFeMnNi})_{60}\text{Ti}_{64}_{40}$



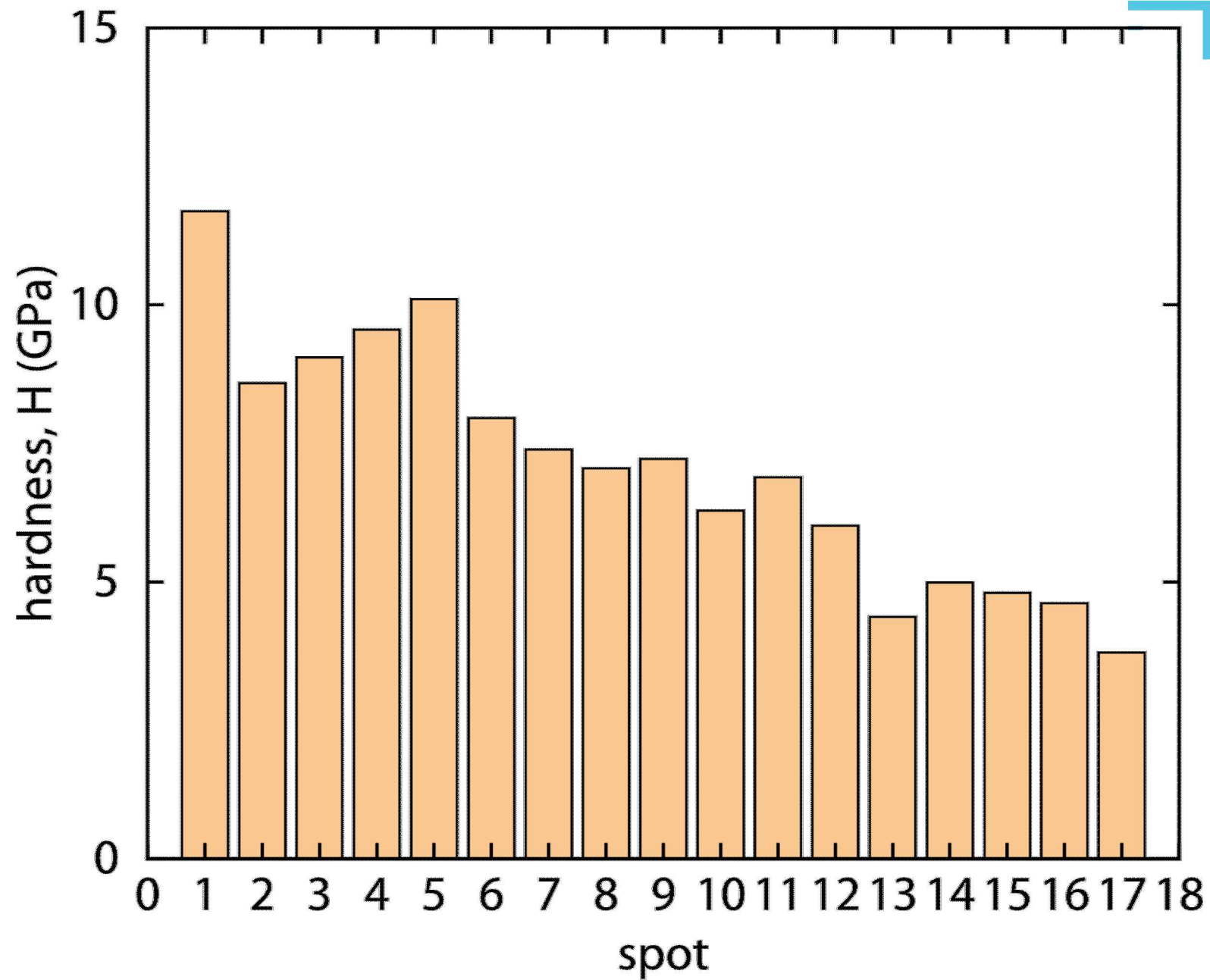
3)  $(\text{CoCrFeMnNi})_{95}\text{Ti}_{64}_{5}$



Microstructural characterization along this sample helped refine composition selections to ones that exhibit chemical homogeneity, expected to be a requirement for the Cr and Ti to behave synergistically in corrosive environments.

Equiatomic  
CoCrFeMnNi

spot	hardness (GPa)
1	11.7
2	8.6
3	9.1
4	9.6
5	10.1
6	8.0
7	7.4
8	7.0
9	7.2
10	6.3
11	6.9
12	6.0
13	4.4
14	5.0
15	4.8
16	4.6
17	3.7





hardness (GPa)

$$H_s = \frac{8F_n}{\pi w^2}$$

$$K_c = \frac{F_t}{(2pA)^{1/2}}$$

$$A = \frac{1}{2} R l_p - (R - d_p) \sqrt{R^2 - (R - d_p)^2}$$

$$l_p = 2R \arccos \frac{R - p_d}{R}$$

$K_c$  is fracture toughness

$F_t$  is friction force, measured and recorded during scratch

$A$  is projected frontal area

$p$  is maximum penetration depth, SWLI

$l_p$  is perimeter length

$R$  is radius of indenter