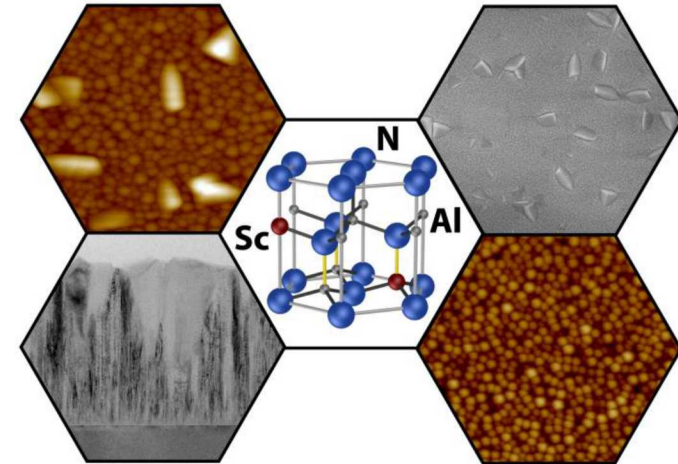
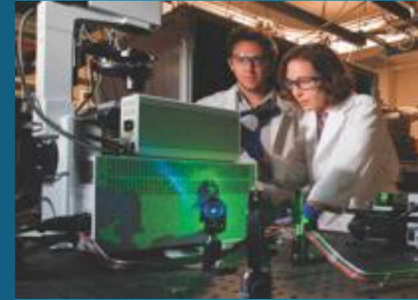




Sandia
National
Laboratories

SAND2020-9582PE

Ultra-Wide Bandgap Materials: Enabling Technologies from Power Electronics to MEMS



PRESENTED BY

Erica A. Douglas

with B. Klein, A. G. Baca, A. A. Allerman,
A. M. Armstrong, R. J. Kaplar, G. Esteves,
M. D. Henry, and B. Griffin



Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.



Power Electronics

Motivation for Ultra-Wide Bandgap Semiconductors

- Material Comparison
- Figure of Merits

Lateral Device Performance

- Depletion Mode
- Temperature Stability (Univ. of Florida Collaboration)
- Enhancement Mode

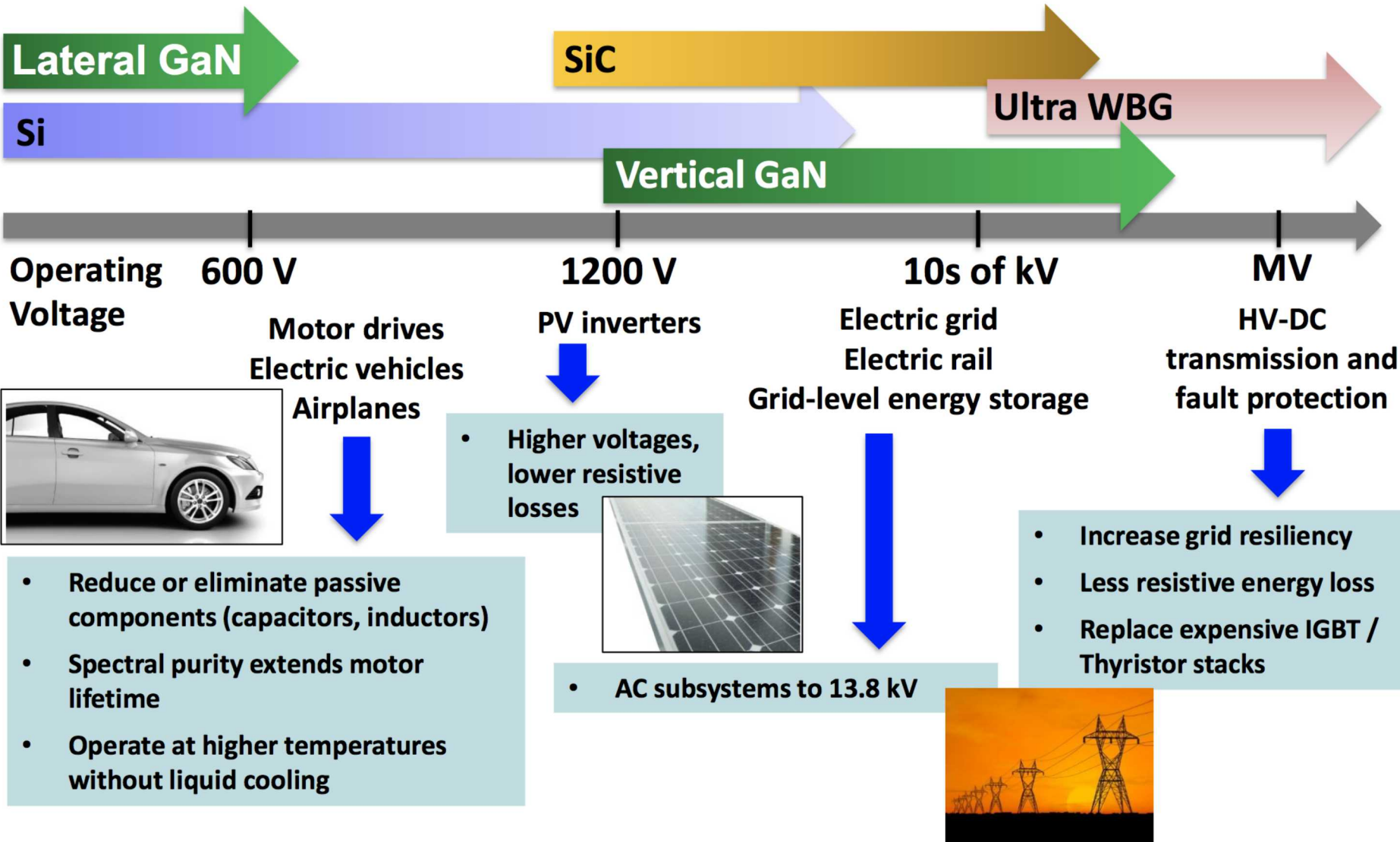
MEMS

Motivation for AlN and alloying with Sc

- Material Challenges
- Device Performance

High Temperature Operation

- High Temperature Performance
- High Temperature Stability

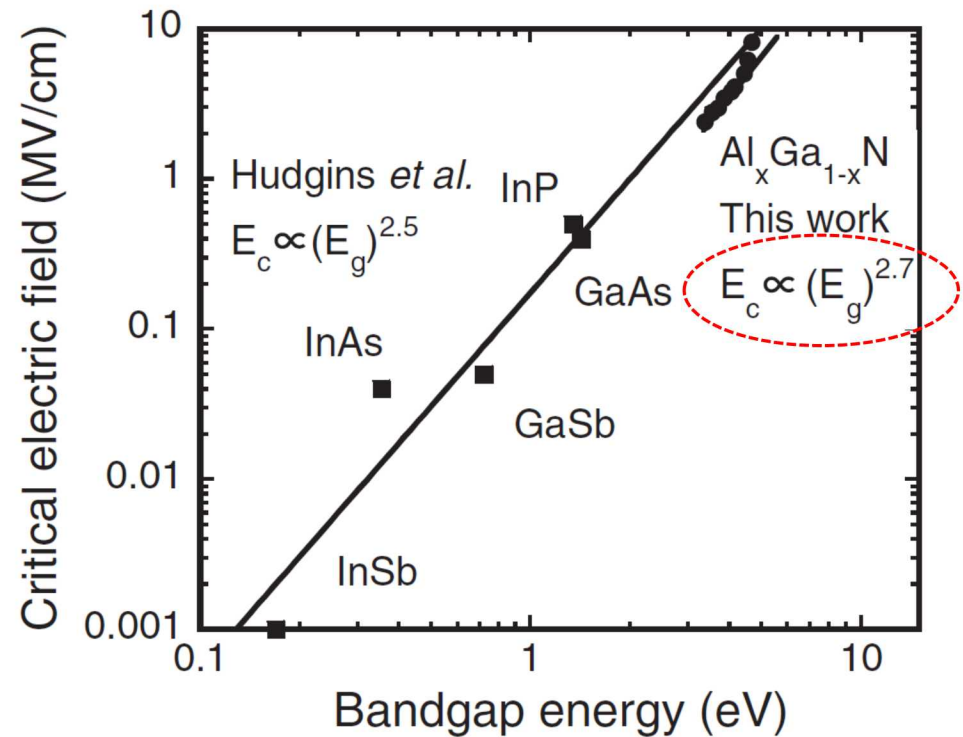




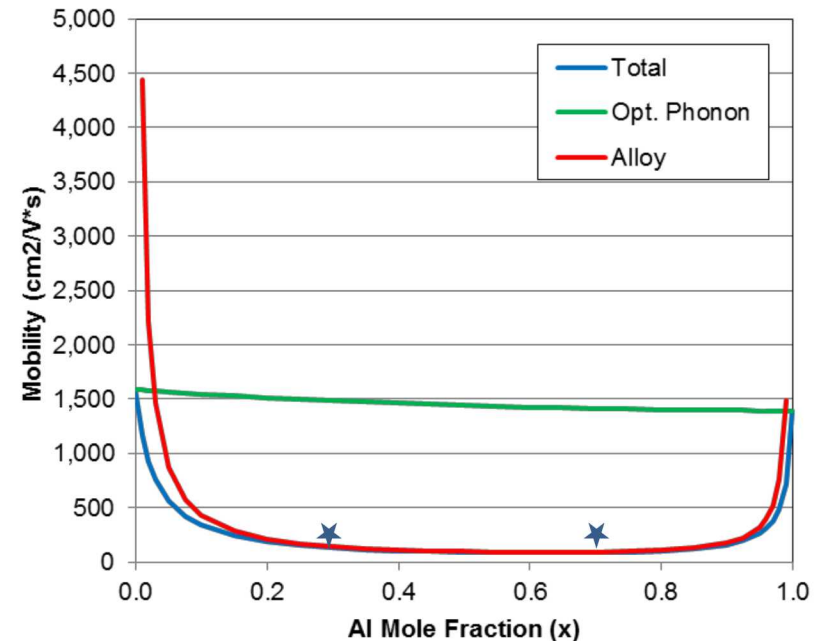
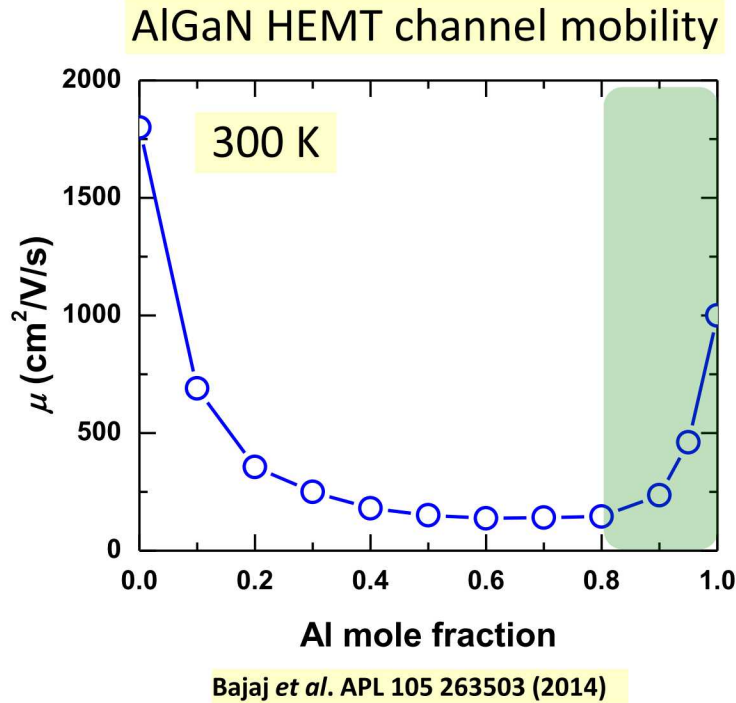
Band gap energy (E_g) is a critical material property for power electronics

$$R_{on,sp} = \frac{V_{br}^2}{q\mu n_s E_{crit}^2}$$

Material	E_{crit} (MV/cm)
Si	0.3
SiC	3.4
GaN	4.0
$Al_{0.3}Ga_{0.7}N$	7.3
β - Ga_2O_3	8
$Al_{0.7}Ga_{0.3}N$	12.7
AlN	17



Nishikawa, *et al.*, JJAP 46, 2316 (2007).

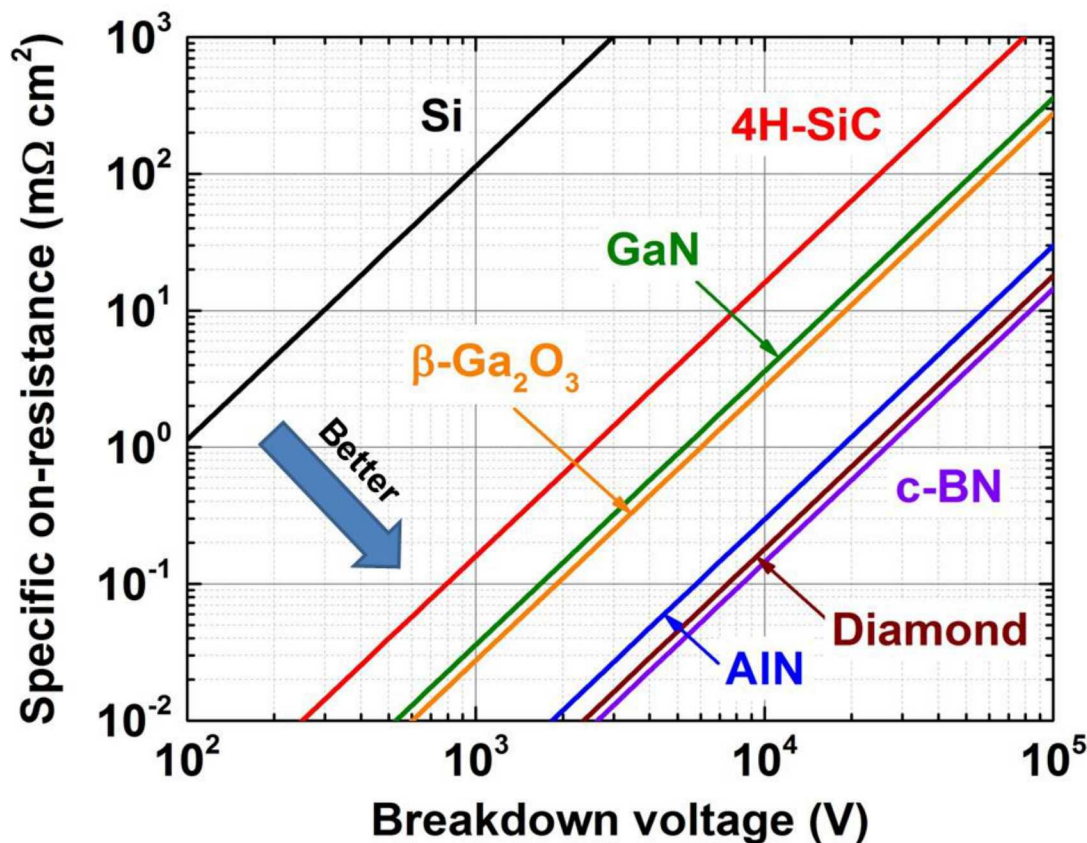


Al_xGa_{1-x}N mobility increases markedly at $x > 0.9$

Binary alloy mobility drops markedly with (extreme) temperature

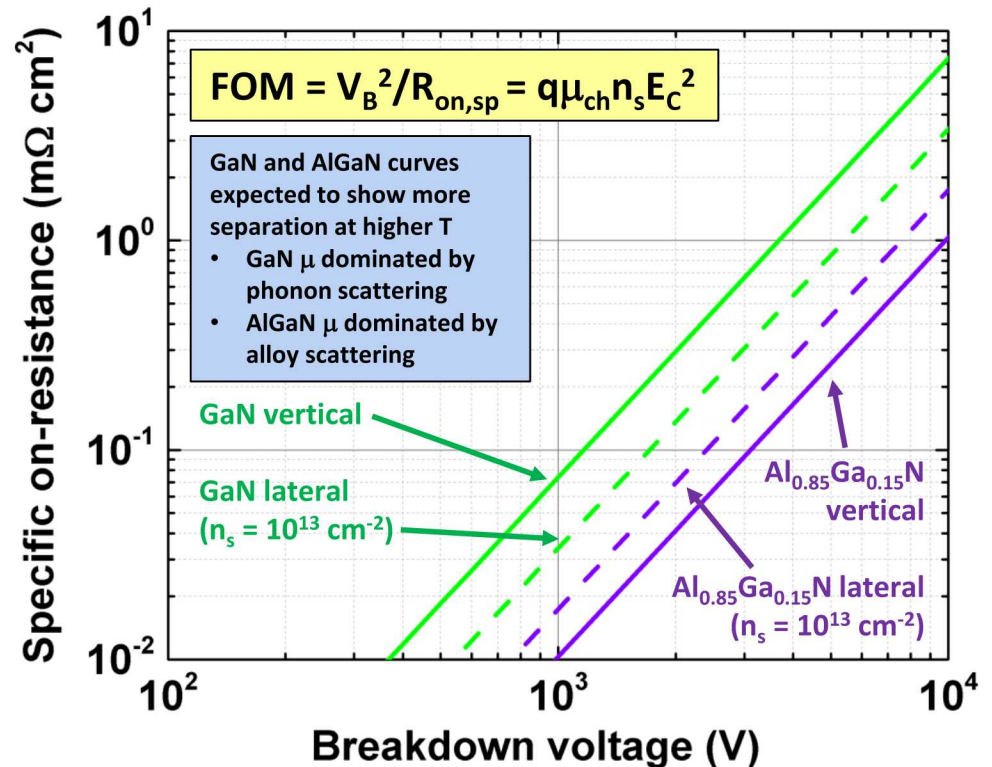
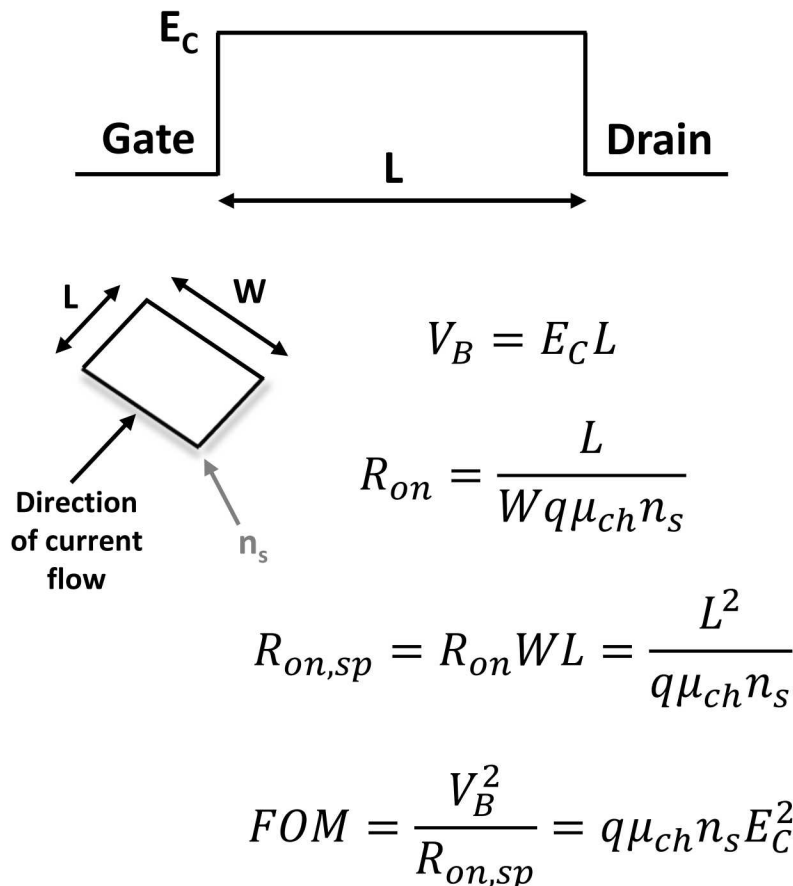


Utilizes refreshed parameters for WBGs and best-known parameters for UWBGs



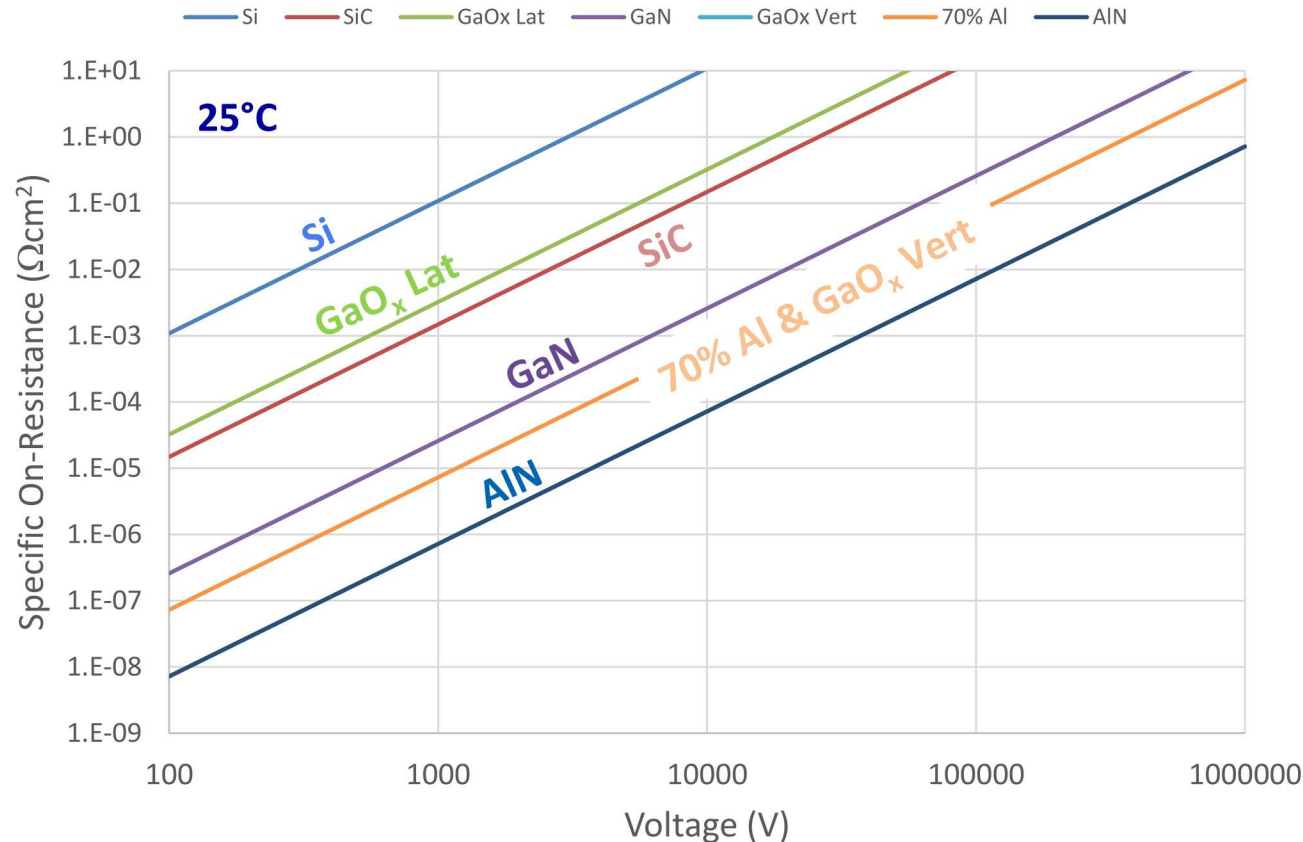
- Diamond, c-BN, and AlN lead the pack
 - c-BN the least mature (difficult to grow in cubic form)
- Caveats:
 - Some material parameters are unknown
 - Some material parameters are not constants
 - Assumes full ionization of dopants
 - Unipolar only

- Not as widely known as the unipolar FOM
- Unipolar (vertical) FOM is often incorrectly used for lateral devices

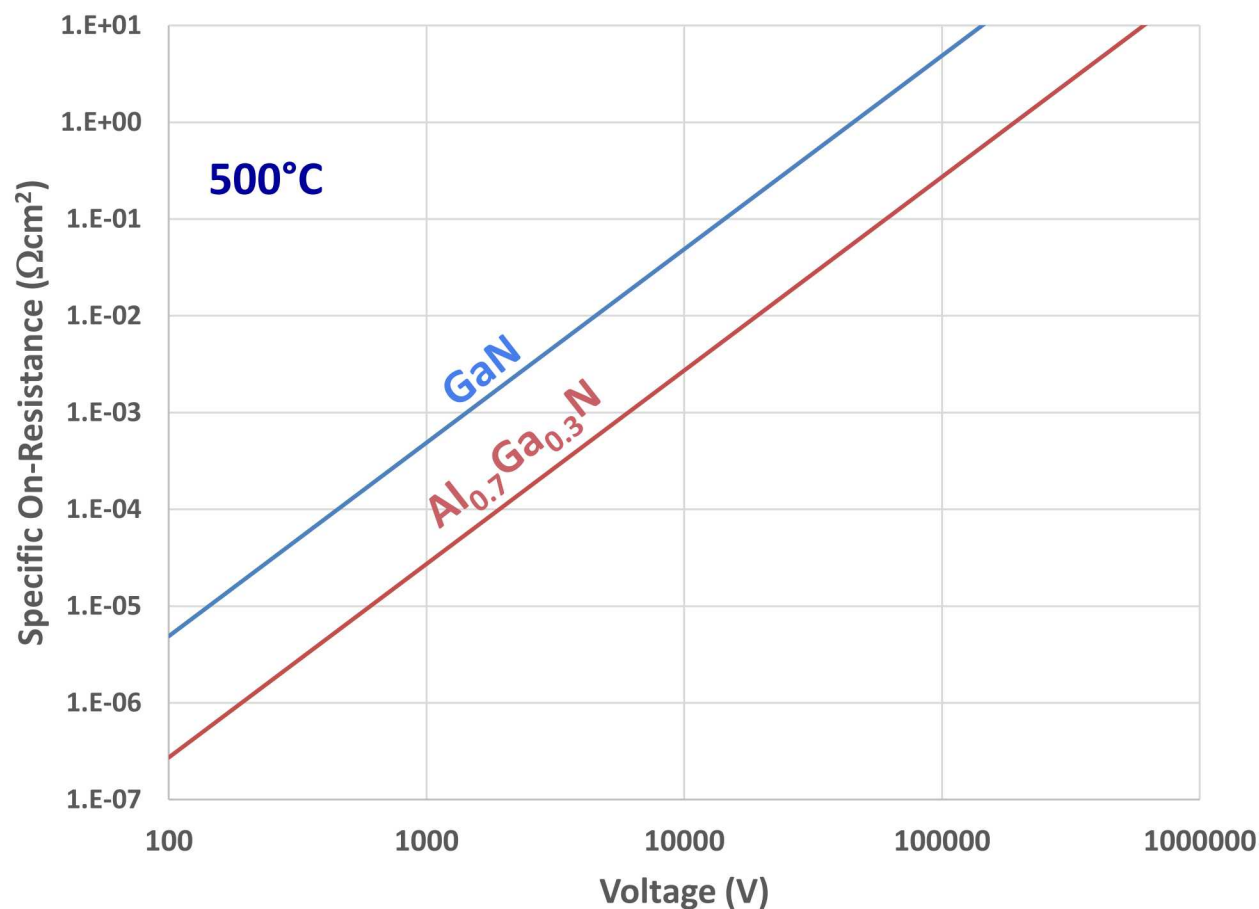


- Proportional to E_C^2 rather than E_C^3 , but high n_s can result in high FOM

Room Temperature Figure of Merit Lateral Comparison



- SiC & GaN are well represented in current research
- Ga_2O_3 is not competitive as a lateral device – path to vertical??
- Everything beyond SiC and GaN is immature
 - Al-rich lateral devices – this work
 - AlN least mature - doping & conductivity issues

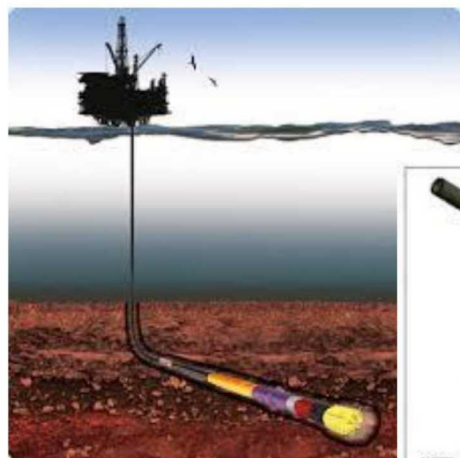


FOM for $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ HEMT is 16x superior to that for GaN at extreme temperature

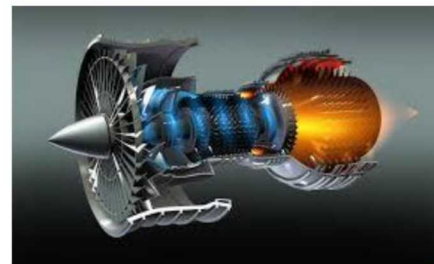


Relevant extreme environments for power electronics:

- Temperature extremes
- Vibration
- Radiation



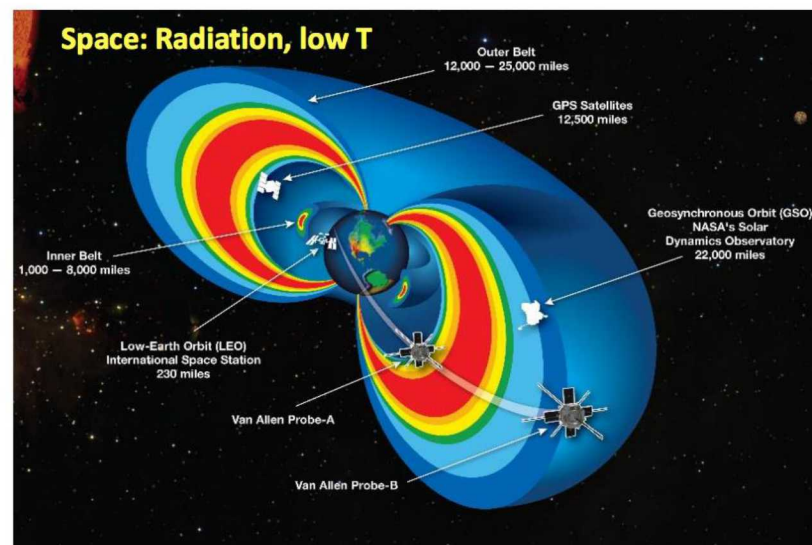
**Down-hole: High T,
vibration**

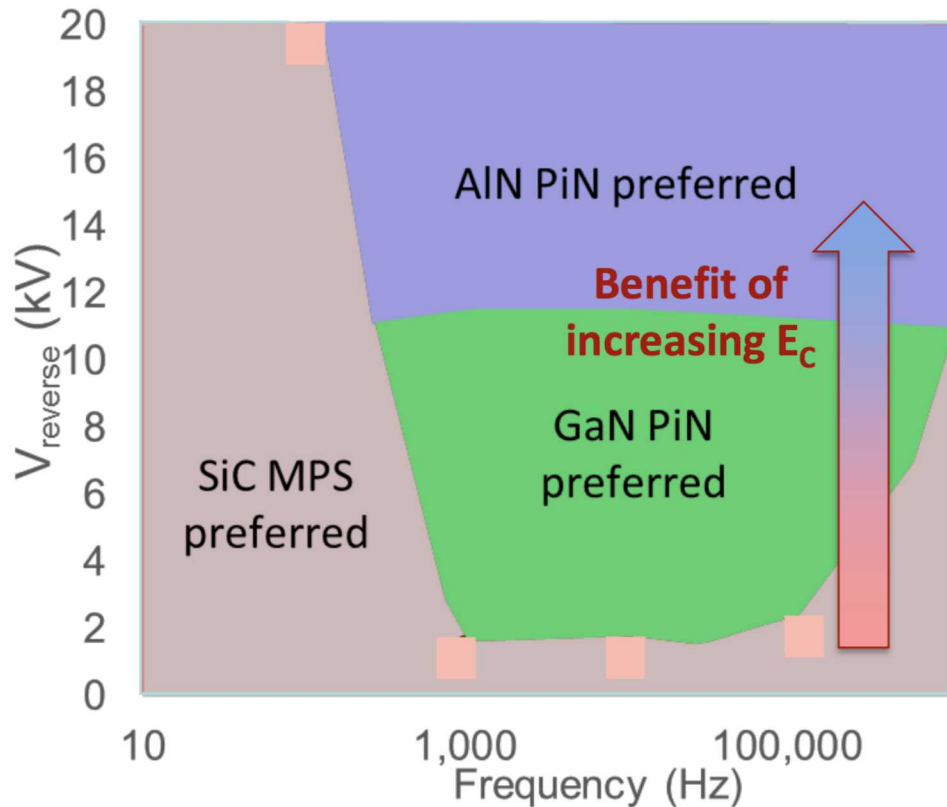


**Aviation: High T,
vibration**

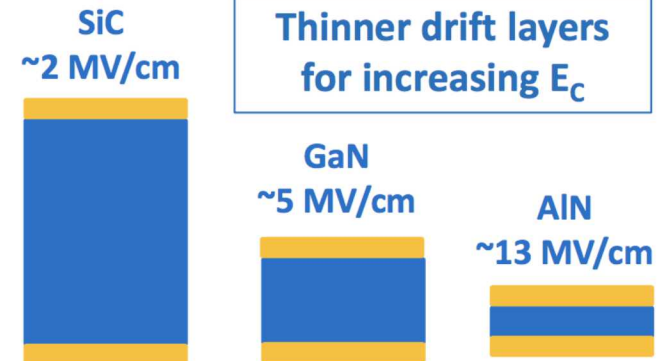


***Avalanche ruggedness is important for
numerous power conversion circuits***



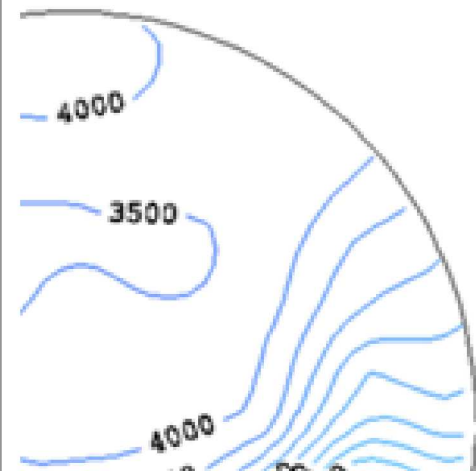
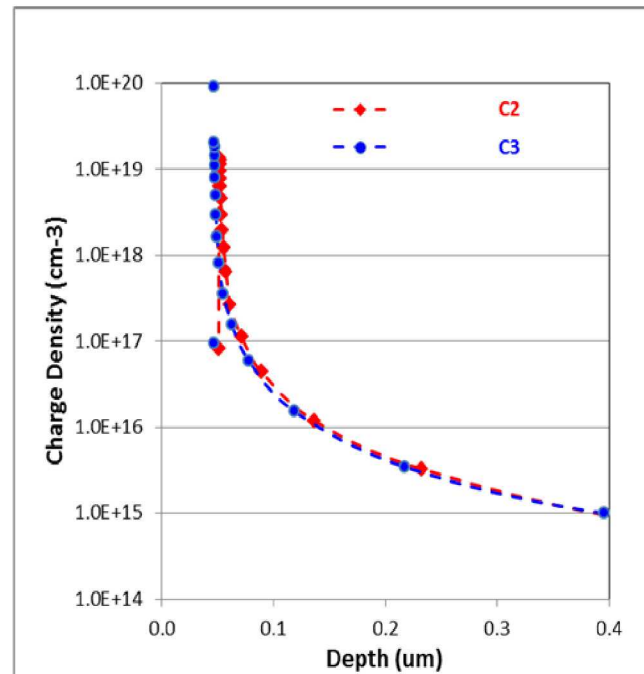
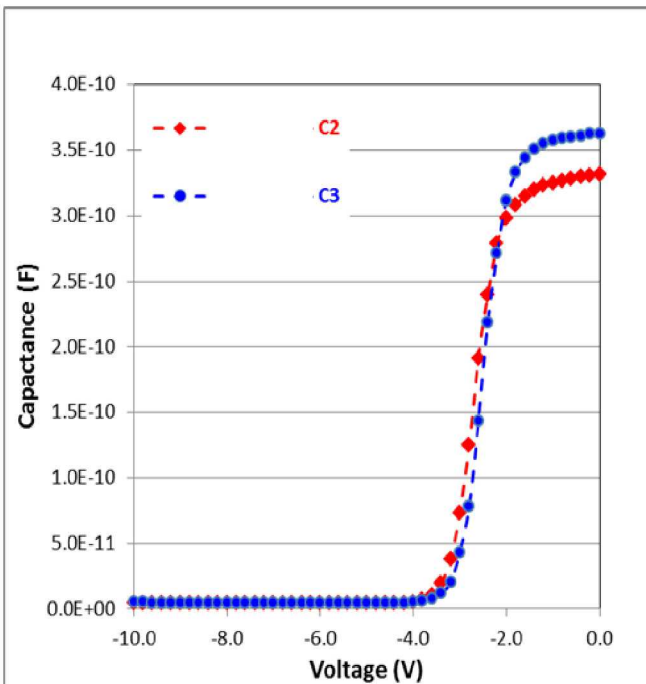
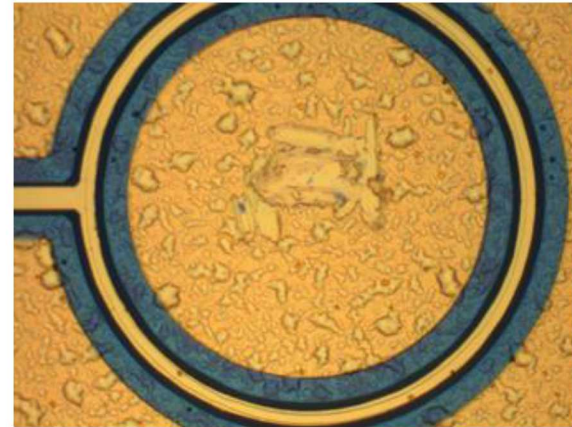
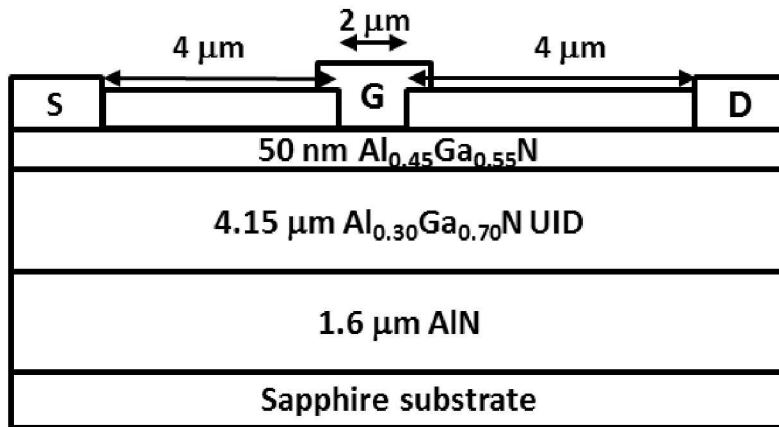


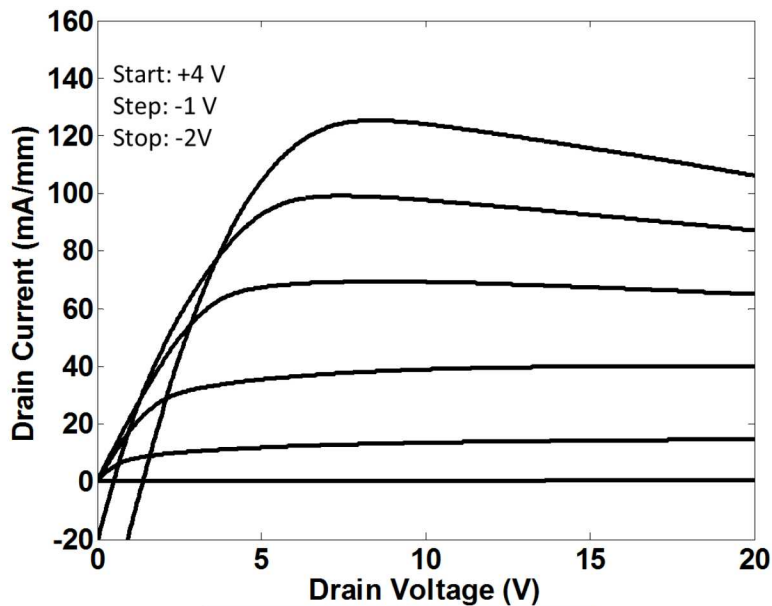
300 K, 50% duty cycle, 500 A/cm²



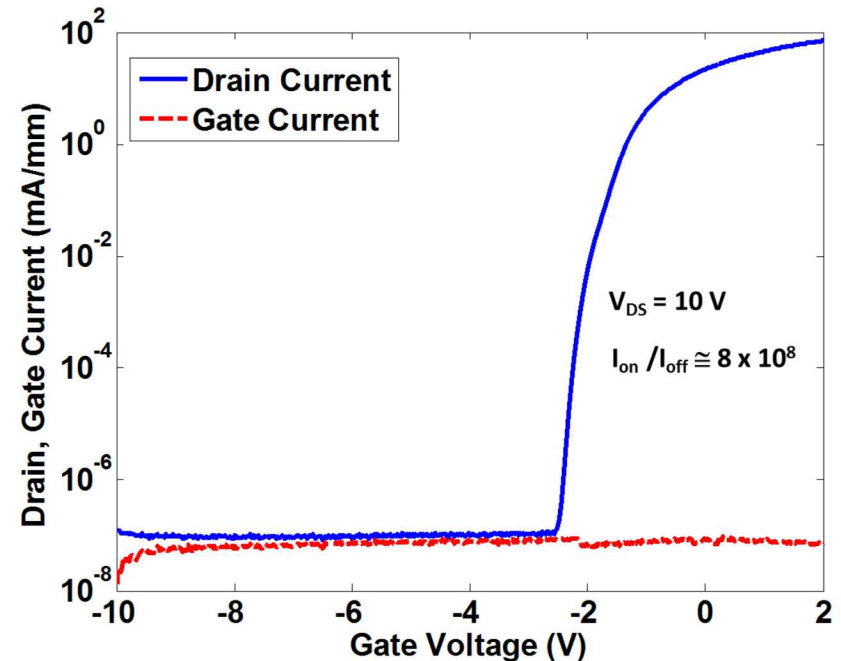
- **GaN and AlN preferred at high voltages over mid-frequency range**
 - Benefit of higher E_c
 - Not as beneficial at low and high frequency (low conductivity modulation and increasing reverse recovery)
- **Examined PiN diodes since peak field is buried below surface**
 - Part of more advanced devices
 - Also must consider Schottky

D-mode $\text{Al}_{0.45}\text{Ga}_{0.55}\text{N}/\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ HEMT



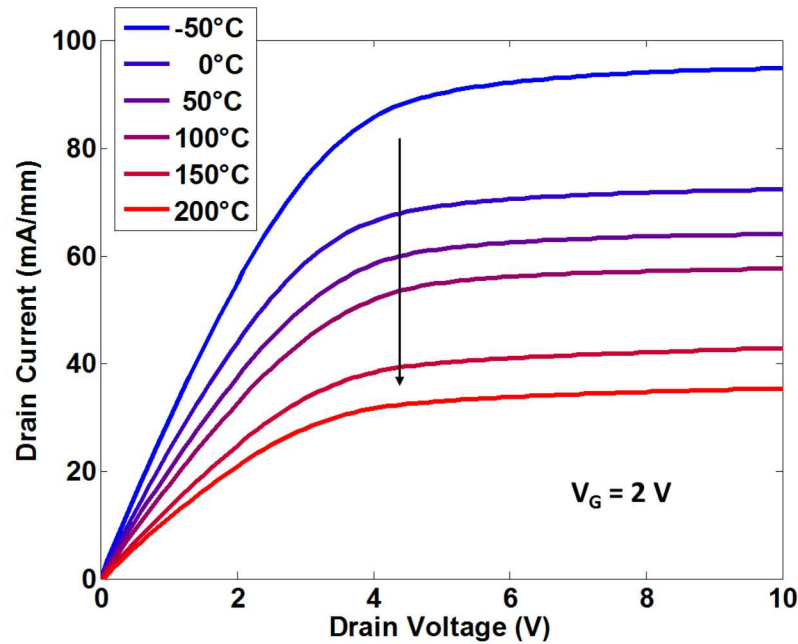


Baca et al. JSSST 6, S3010 (2017).

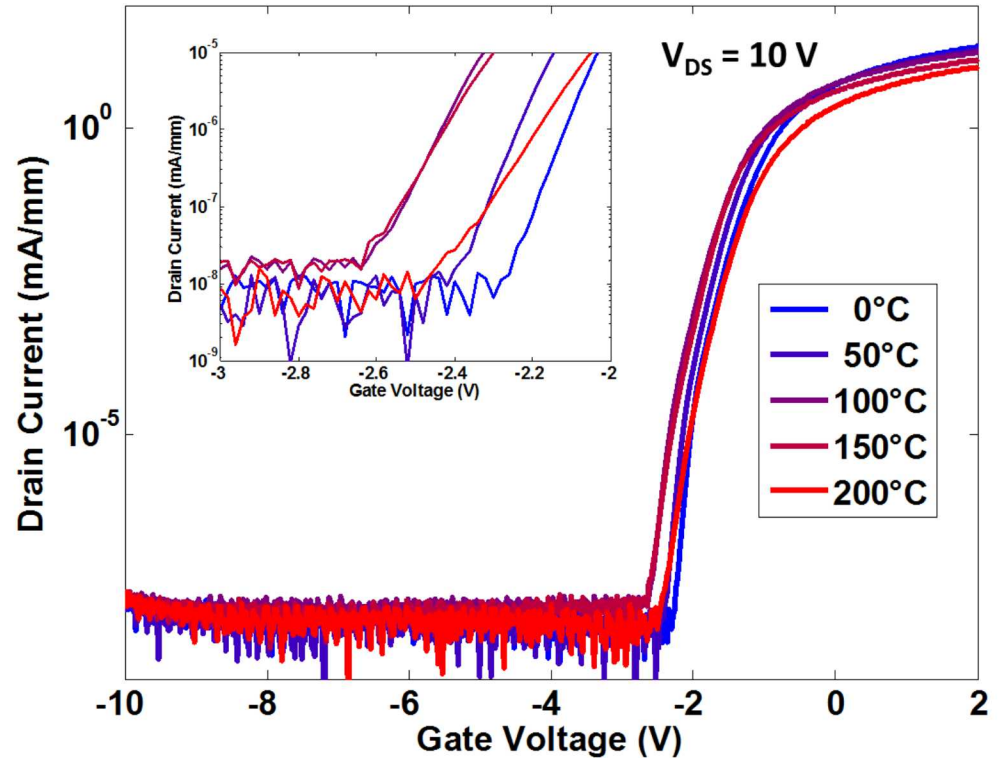


- Linear through the origin I-V due to $\sim 2.5 \times 10^{-5} \Omega\text{-cm}^2$ Ohmic contacts
- Exceptional $I_{D\text{MAX}}/I_{D\text{MIN}}$ ratio (I_{on}/I_{off}) of 8×10^8
- Very low gate leakage

Temperature Dependence of “45/30” Transfer and Output Characteristics

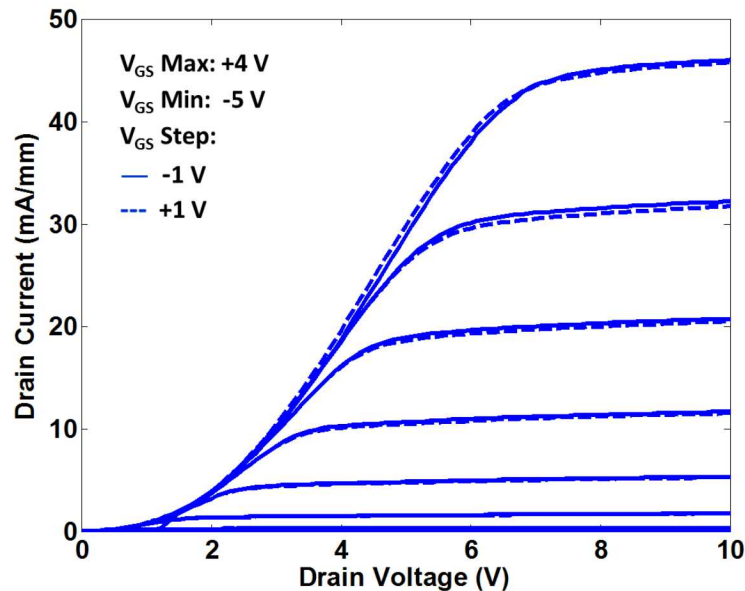
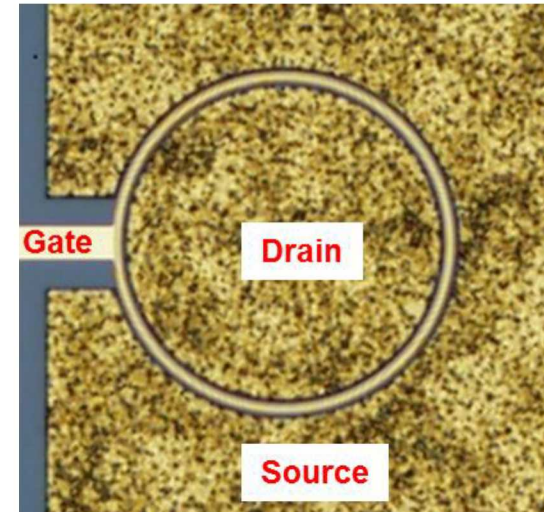
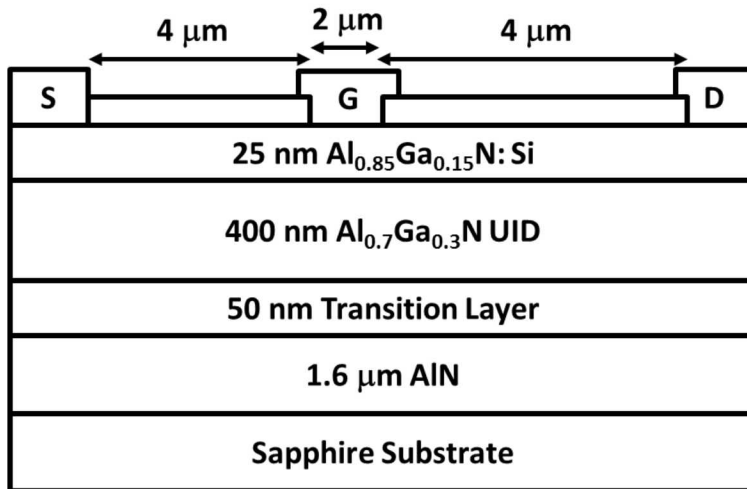


Baca et al. JSSST 6, S3010 (2017).



- Drain current decreases monotonically with temperature
- Exceptionally low leakage current is invariant with temperature
- Subthreshold slope varies slightly with temperature

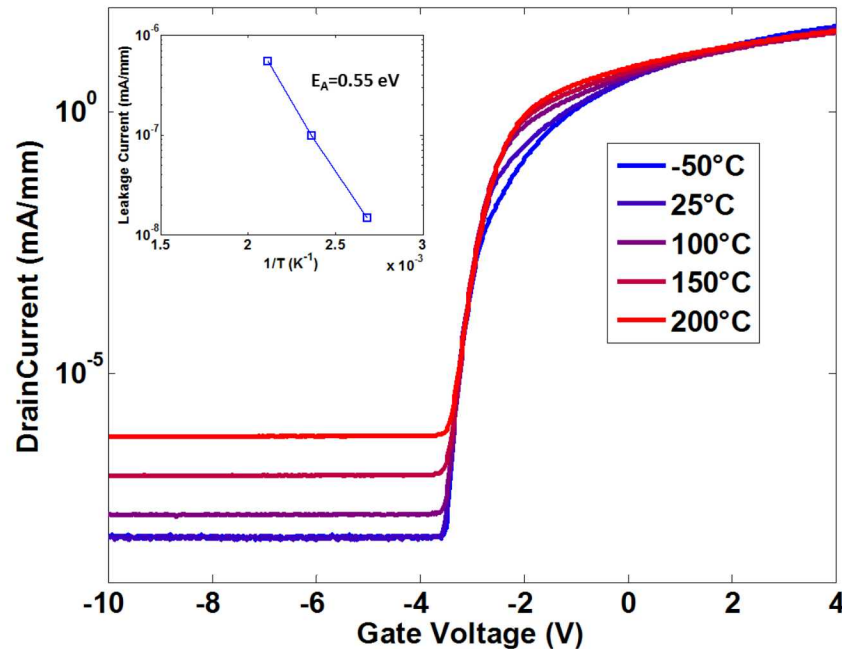
D-mode $\text{Al}_{0.85}\text{Ga}_{0.15}\text{N}/\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ HEMT



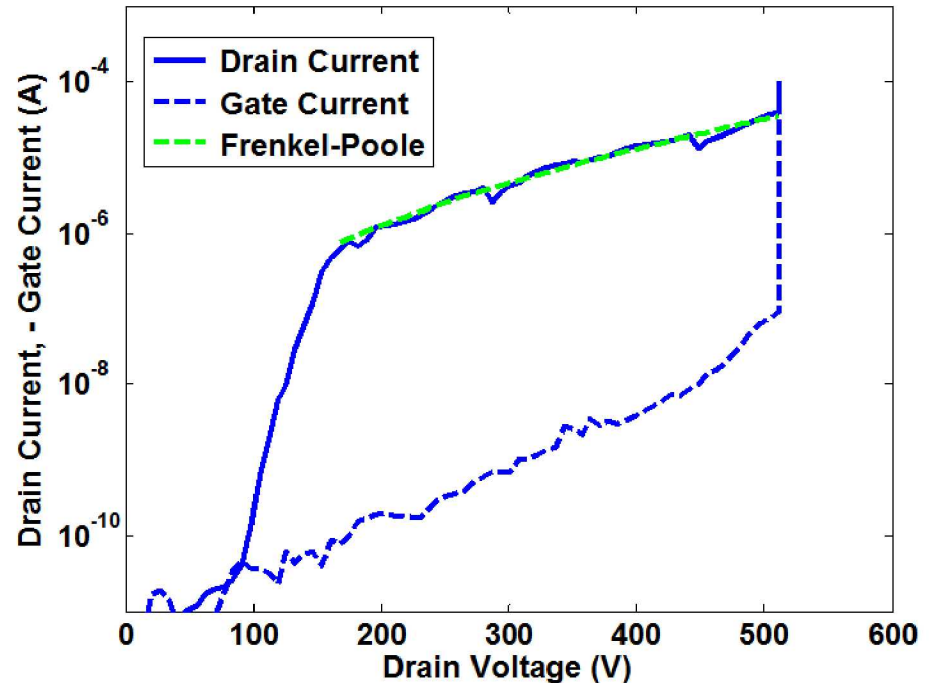
Baca et al. JSSST 6, Q161 (2017).

- Very little hysteresis
- Greater difficulty in Ohmic contact formation leads to offset voltage and further limitation of drain current potential

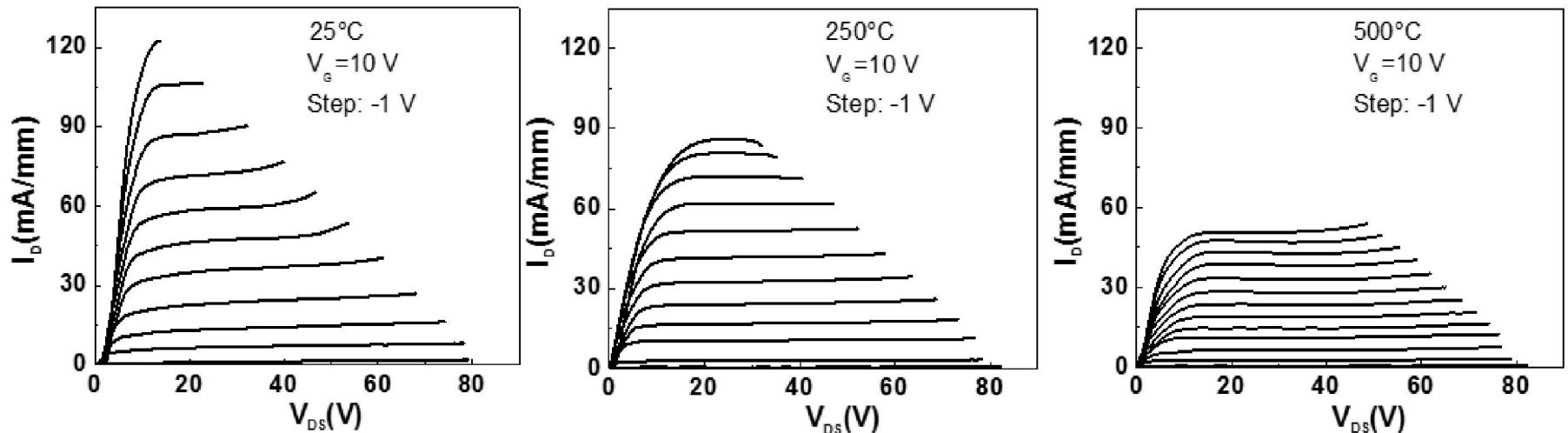
Exceptional Leakage Characteristics over Temperature in D-mode “85/70” HEMT



Baca *et al.* JSSST 6, Q161 (2017).



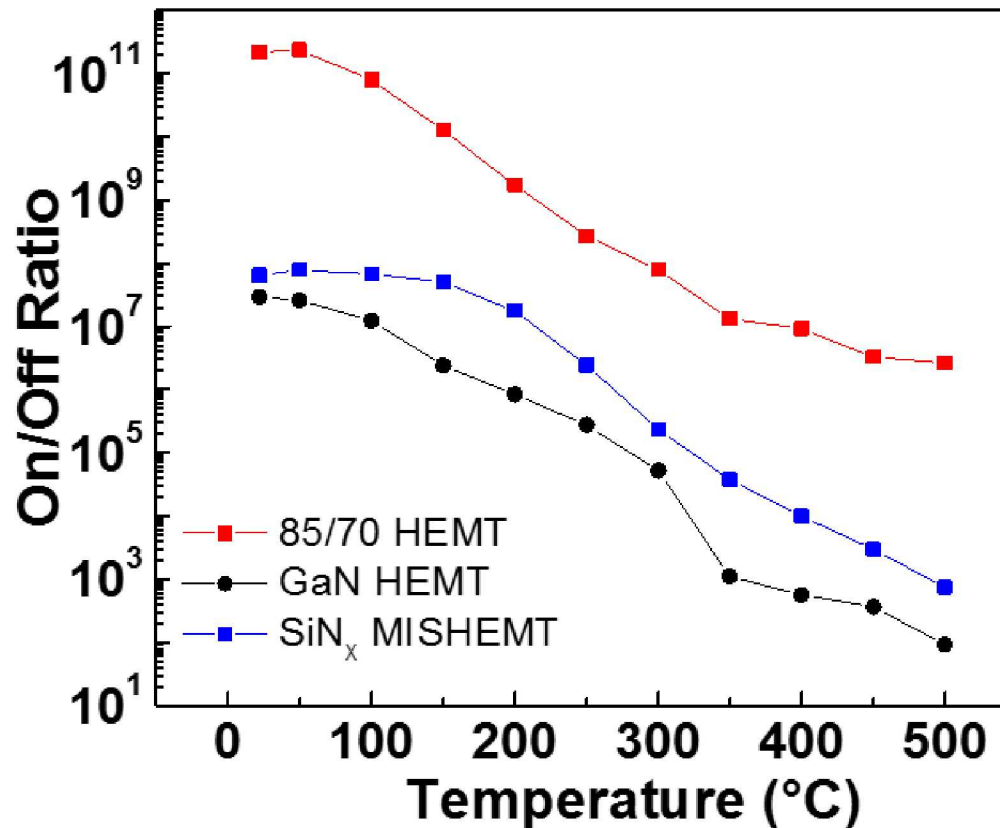
- Exceptional ratio I_{on}/I_{off} ratio of 8×10^9 at -50°C
- Exceptionally low leakage current that increases with temperature and $E_A = 0.55$ eV
- Breakdown voltage exceeds 500 V and drain leakage current is consistent with Frenkel-Poole conduction



Carey *et al.* submitted to IEEE Trans. El. Dev.

- Current density exceeds 120 mA/mm from increasing the forward gate bias
- Retain excellent current modulation and over 50 mA/mm current density when operated at 500°C

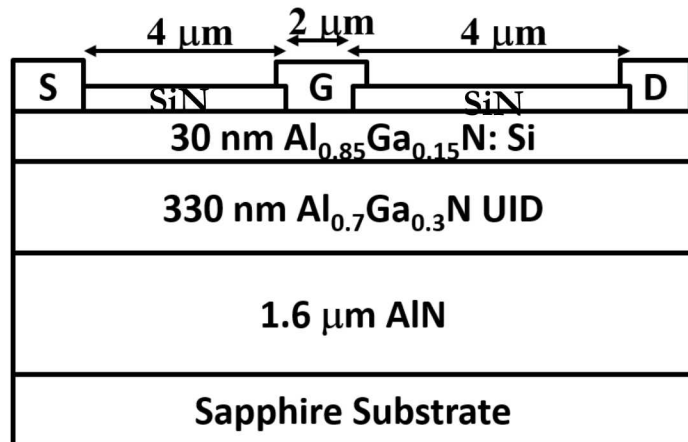
On/Off Ratio of D-mode “85/70” HEMT is Favorable at High Temperature Operation



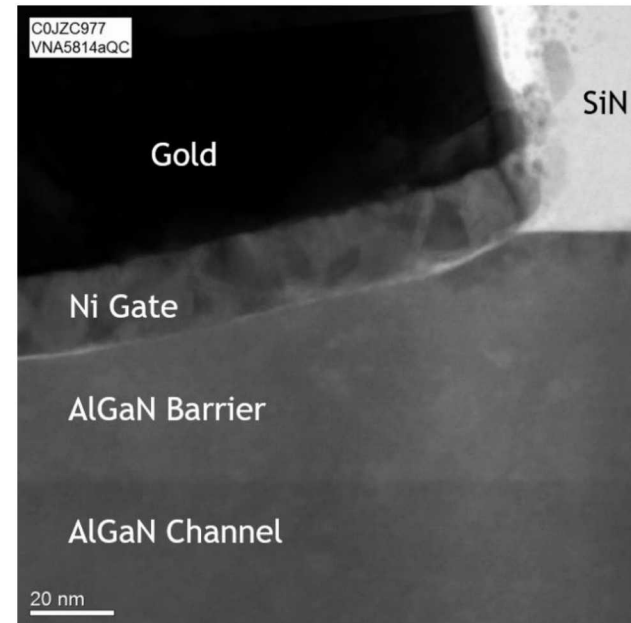
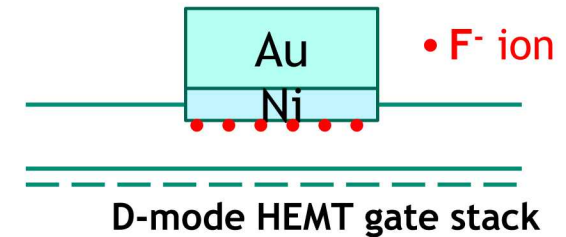
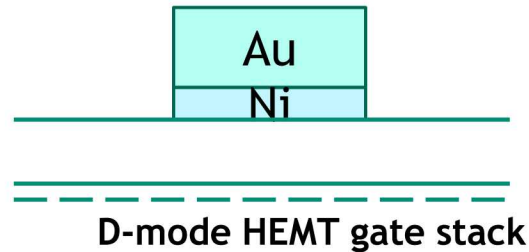
Carey *et al.* submitted to IEEE Trans. El. Dev.

- Al-rich HEMT maintains excellent On/Off ratio at 500°C and GaN doesn't
- On/Off ratio is fairly flat from 200-500°C

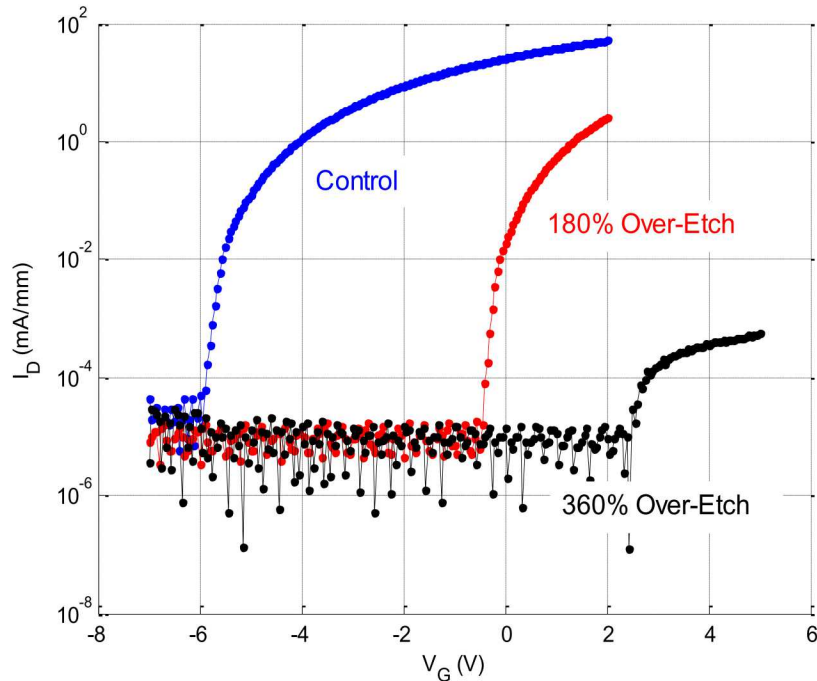
E-mode $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ using *F-Plasma* Treatment



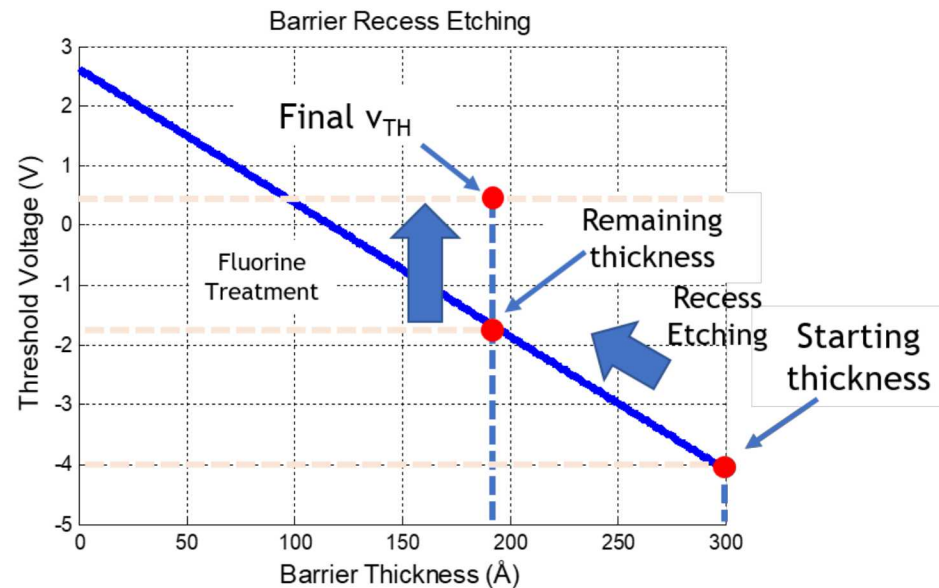
Motivated by Cai *et al.*, IEEE Trans. El. Dev. 53, 2223 (2006)



F-plasma from reactive ion etch (through SiN): combination of AlGaN etch and incorporation of F^- ions near the surface



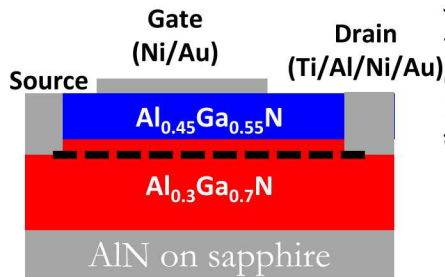
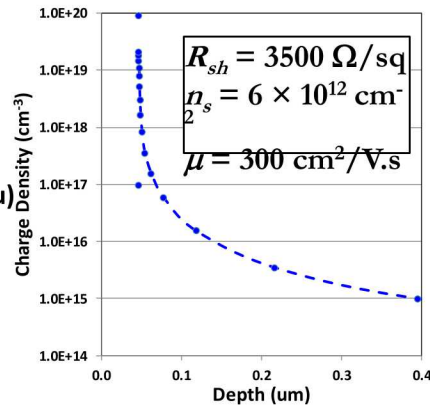
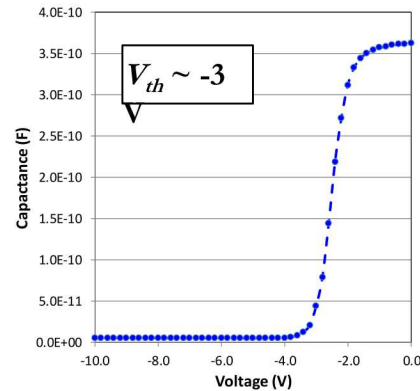
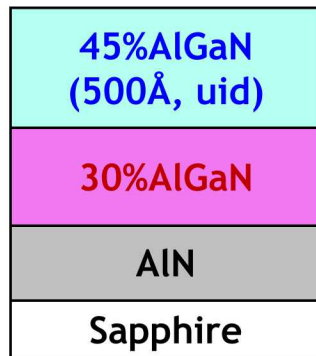
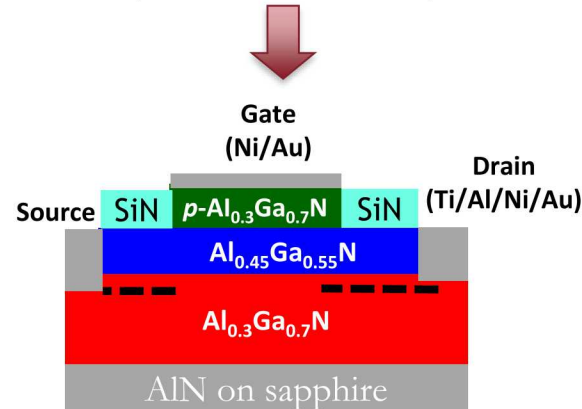
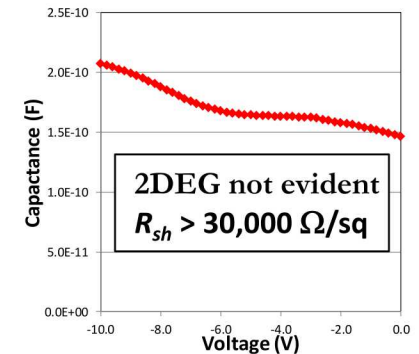
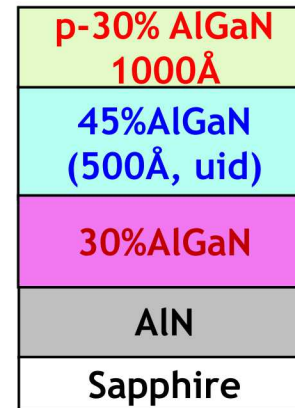
$$V_{\text{TH}} = \varphi_b - \frac{q\sigma_p t_{\text{barrier}}}{\epsilon_r \epsilon_0} - \Delta E_c$$



F-plasma from reactive ion etch (through SiN, then into AlGaN): combination of AlGaN etch and incorporation of *F* ions near the surface



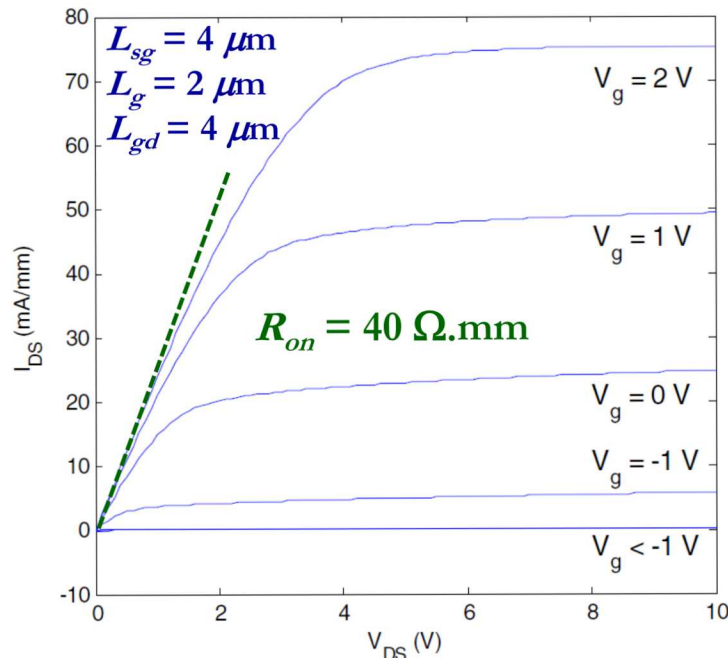
D-mode 45/30 HEMT

E-mode *p*-30/45/30 HEMT

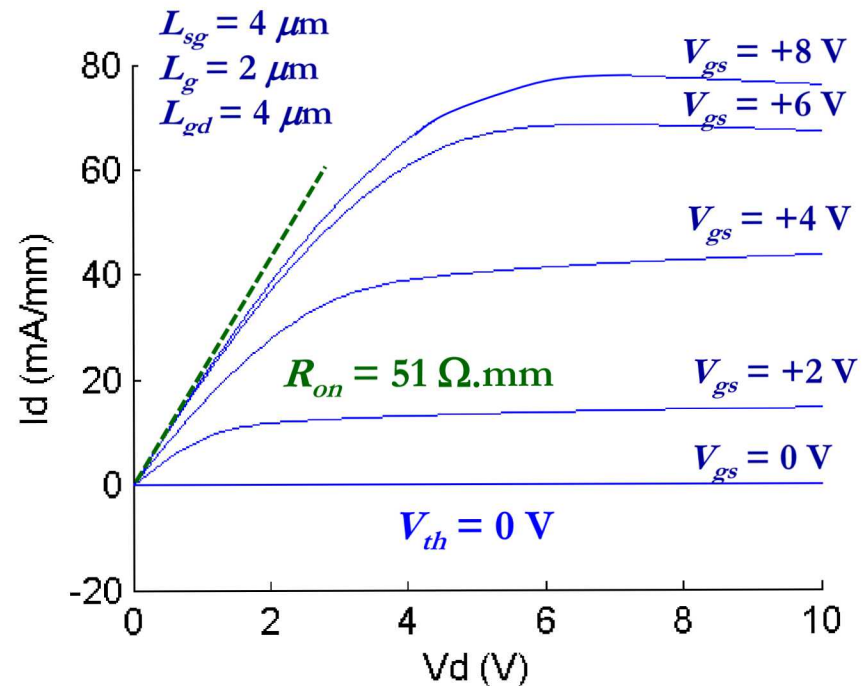
- MOVPE-grown UID Al_{0.45}Ga_{0.55}N/Al_{0.3}Ga_{0.7}N (**45/30**) on Al_{0.3}Ga_{0.7}N-on AlN-on sapphire
- *p*-Al_{0.3}Ga_{0.7}N/Al_{0.45}Ga_{0.55}N/Al_{0.3}Ga_{0.7}N (***p*-30/45/30**) heterostructures very resistive
- *p*-AlGa_{0.3}N gate defined by ICP etching and passivated with SiN



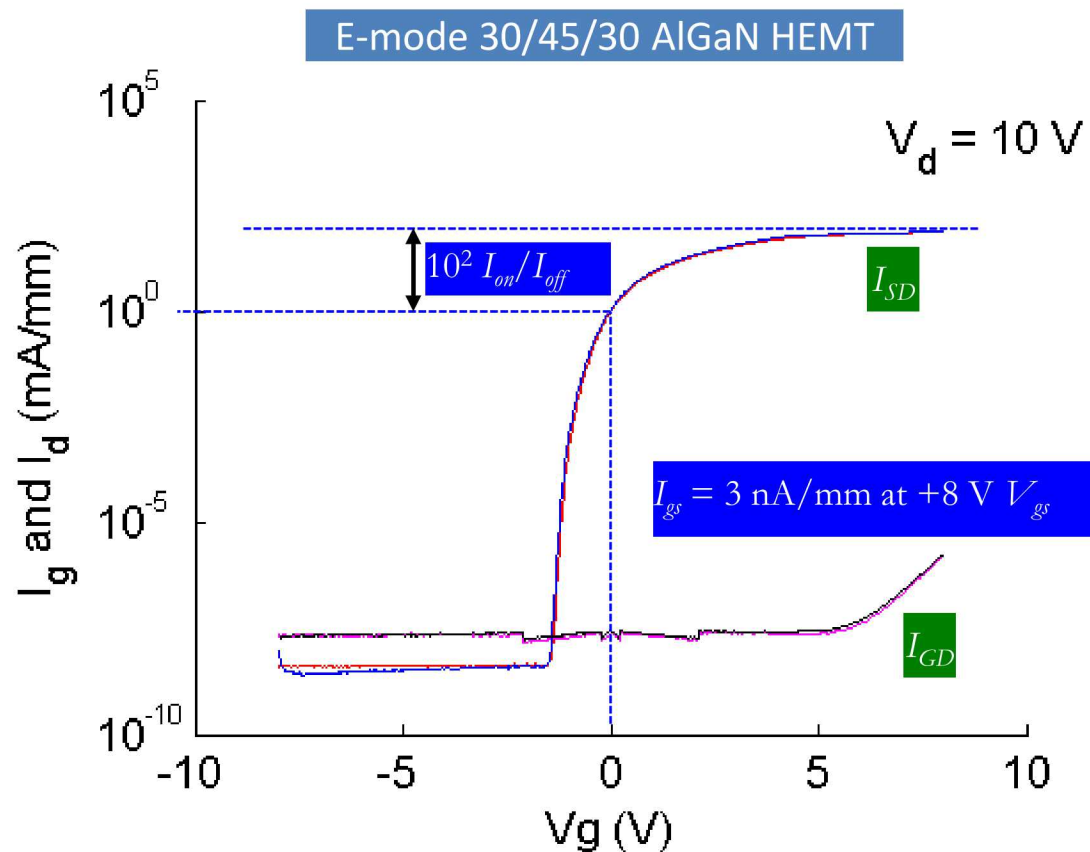
I-V of D-mode 45/30 AlGaIn HEMT

Klein *et al.*, ECS JSSST 6 S3067 (2017).

I-V of E-mode 30/45/30 AlGaIn HEMT

Douglas, *et al.*, DRC, Santa Barbara, CA, June 25, 2018

- $V_{th} = 0 \text{ V}$
- Little to no degradation of I_{max} or R_{on} for E-mode vs. D-mode AlGaIn HEMT



- Small hysteresis ($\sim 80\text{ mV}$)
- Vanishingly small gate current

UWBG Development to Enable MEMS



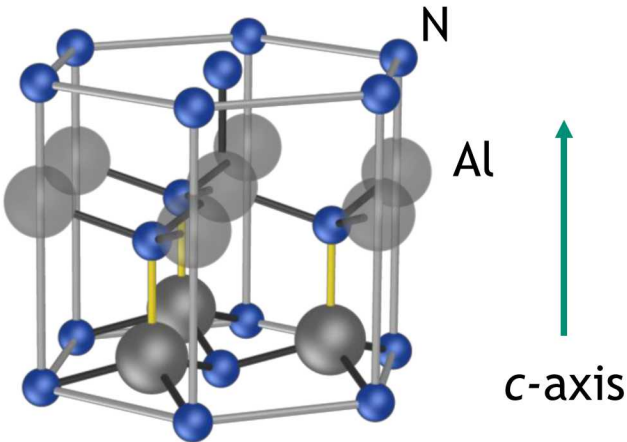
Piezoelectric

- Material can couple electrical and mechanical energy
- Direct effect
 - Converse effect
- Magnitude characterized by piezoelectric constant (d_{im})
- Efficiency characterized by electromechanical coupling coefficient (k_t)

Aluminum Nitride (AlN)

- Piezoelectric film
- d_{33} = 5.5 pC/N
 - d_{31} = -2.5 pC/N
- Reactive sputter deposition
- Low temperature deposition ~350°C
- CMOS compatible

Direct Effect	Converse Effect
Electrically polarize in response to an applied stress	Mechanically strain in response to an applied electric field
$P_i = d_{im} \cdot \sigma_m$	$\varepsilon_m = d_{im} \cdot E_i$

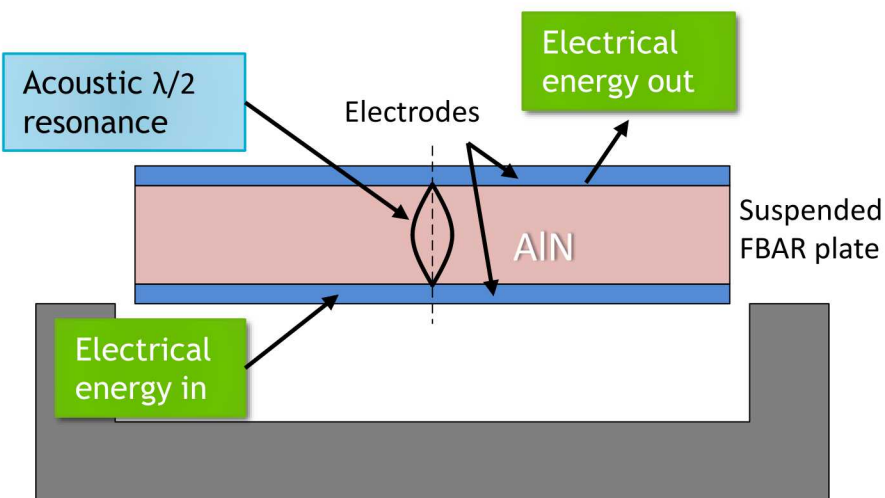


Hexagonal Wurtzite
 a = 3.144 Å
 c = 5.046 Å



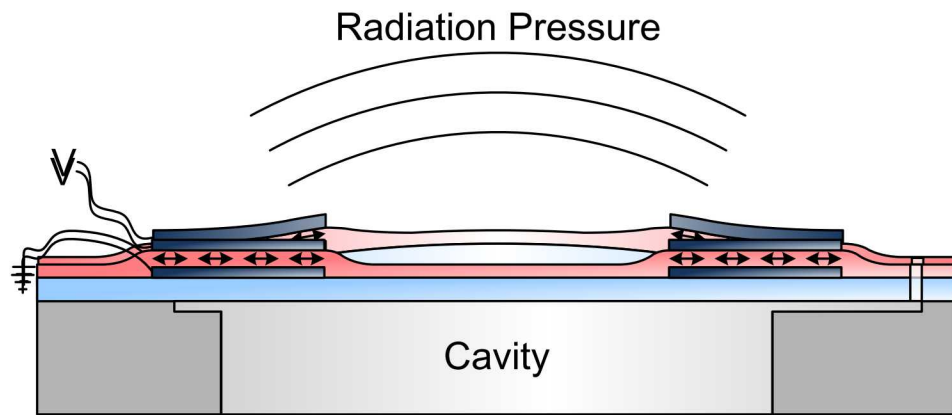
Film Bulk Acoustic Resonator (FBAR) Filters

- Half-wavelength, thickness mode resonators
- Filter frequency is set by film thickness
- Commercial success for AlN MEMS
 - >1 billion units produced per year from Avago Technologies
- More than 40 discrete filter die in the modern smartphone



Piezoelectric Micromachined Ultrasonic Transducer (PMUT) Applications

- Proximity sensing
- Short range communication
- Extreme high sound pressure level microphones
- Imaging



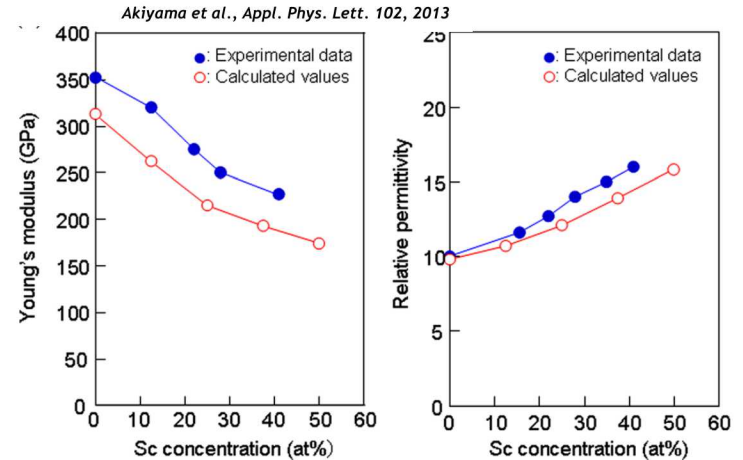
Piezoelectricity

- Material can couple electrical and mechanical energy

ScAlN has the potential to replace current AlN technologies

Alloying Sc into AlN increase piezoelectric response

Challenges in depositing high Sc content films arise and has created a hurdle in current LDRD



Device Performance Based on Properties

Resonators

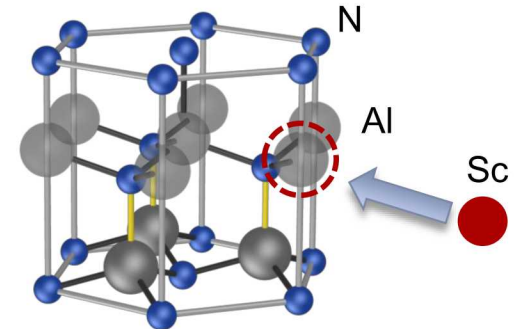
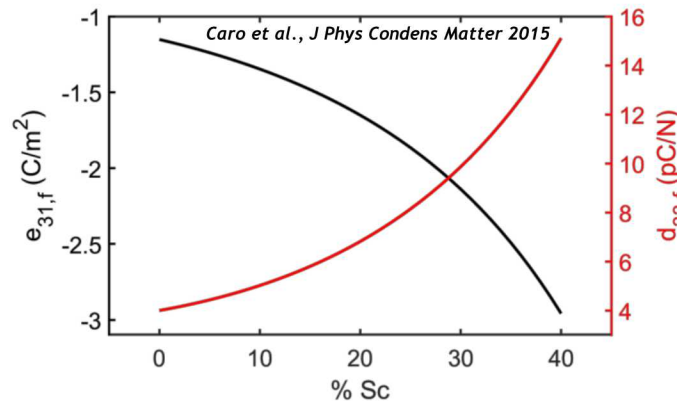
$$k_t^2 = \frac{e_{31}^2}{E_p \epsilon_{33}}$$

Energy harvesting

$$FOM = \frac{(e_{31,f})^2}{\epsilon_{33,f}}$$

Sensors

$$\frac{S}{N} = \frac{e_{31,f}}{\sqrt{\epsilon_0 \epsilon_{33,f} \tan \delta}}$$





Secondary grain growth (inclusions) in ScAlN

- Increase with Sc concentrations
- Density controlled by processing parameters

Inclusions have a diffuse {100} orientation

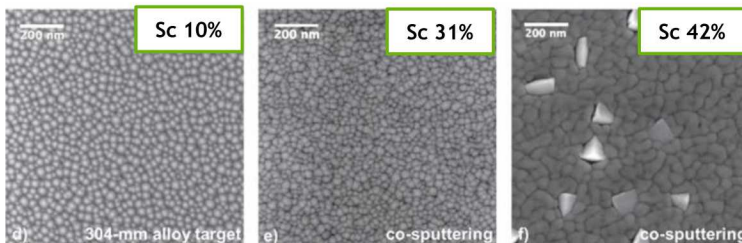
- Fichtner *et. al.*, 2017

Compressive stress suppresses inclusions

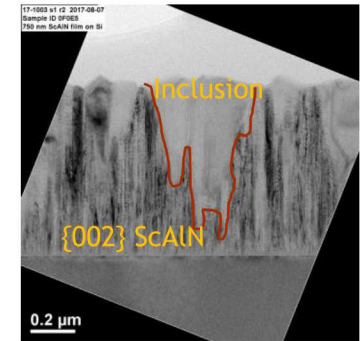
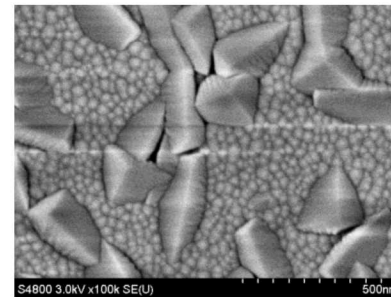
- Henry *et. al.*, 2018
- *Compressive stress not optimal for released MEMS*

Platinum as a bottom electrode on 8" wafers

- Mertin *et al.*, 2018
- Full suppression of inclusions up to Sc 31%
- *Pt is not CMOS compatible and presents integration challenge*

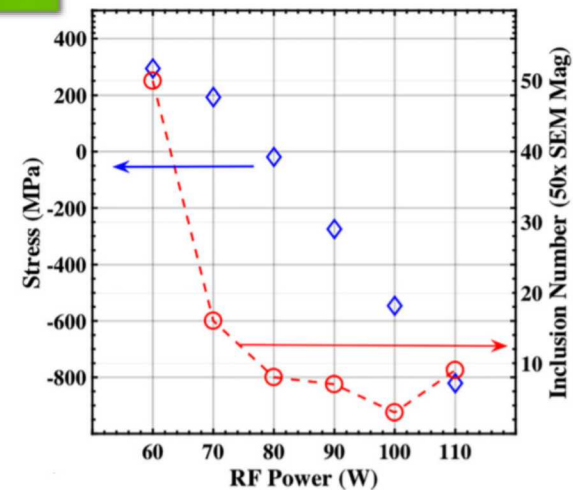


S. Mertin *et al.*, *Surf. Coat. Technol.* (2018)



This work

M. D. Henry *et al.*, *J. Vac. Sci. Technol. B* 36 (2018)

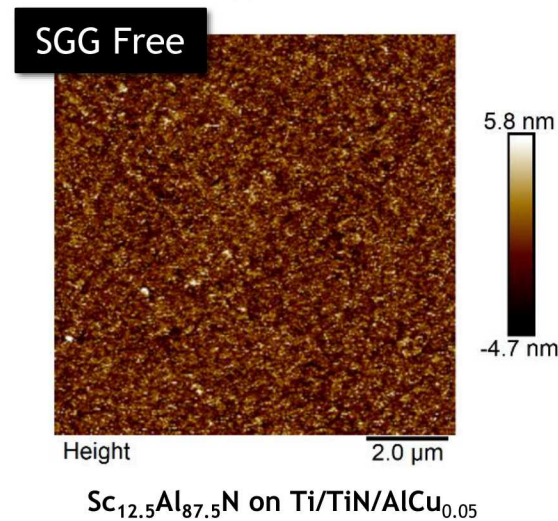
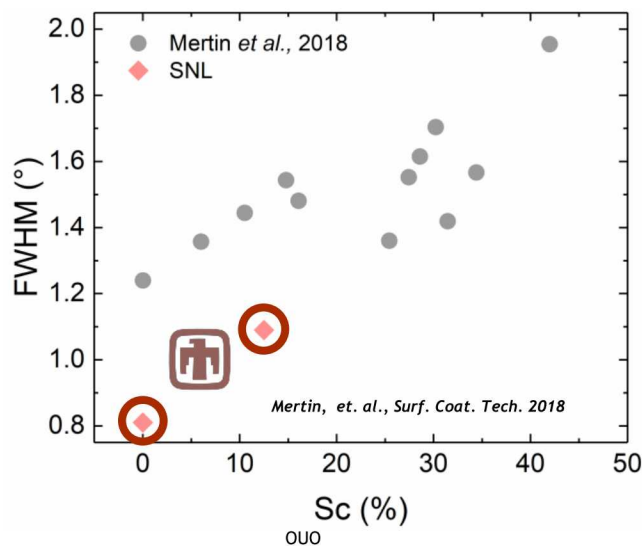
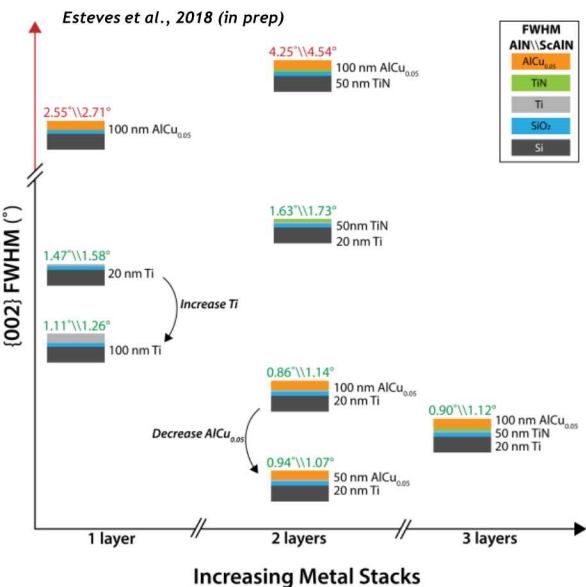
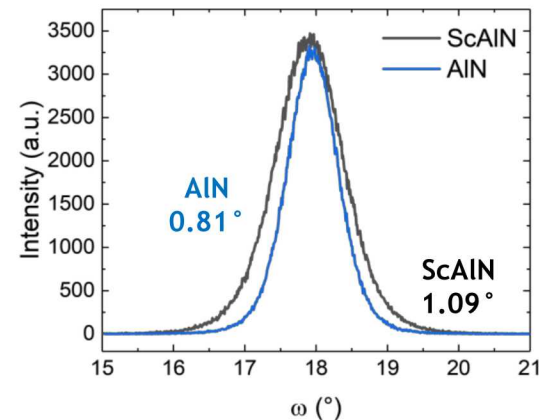


High-quality ScAlN has been deposited on various CMOS compatible metal stacks

- Depositing ScAlN on CMOS metal is **extremely** valuable to industrial sector
- Attractive to industry leaders and has sparked collaboration

Optimal ScAlN film quality found in Ti, Ti/AlCu_{0.05}, and Ti/TiN/AlCu_{0.05} metal stack

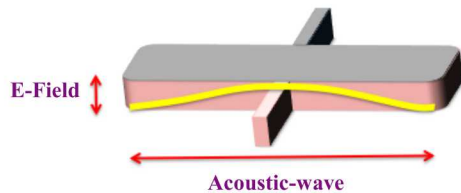
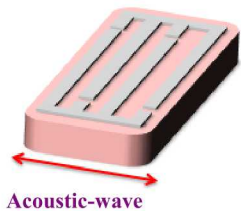
Present results are **state-of-the-art** for their given thickness and Sc composition





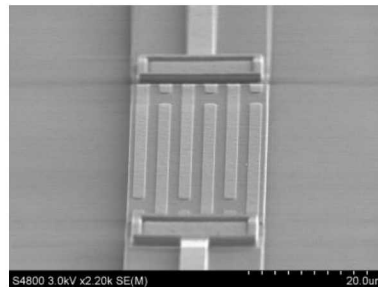
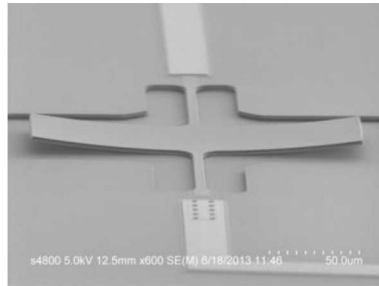
High-quality ScAlN has been deposited on various CMOS compatible metal stacks

- Electric field in z-axis creates an acoustic wave in x-axis via piezoelectric effect
- Device architecture was chosen to show increases in electromechanical coupling with addition of Sc.



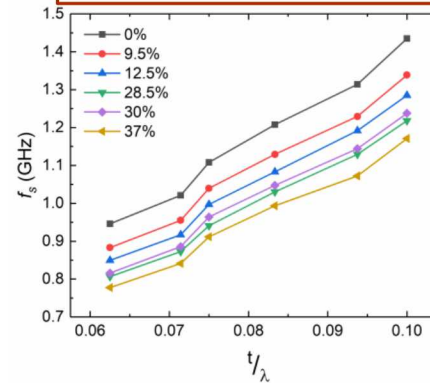
$$w = \frac{n}{2}$$

$$f = \frac{n}{l} = \frac{1}{2w} * \sqrt{\frac{\dot{a} E^* t}{\dot{a} r^* t}}$$

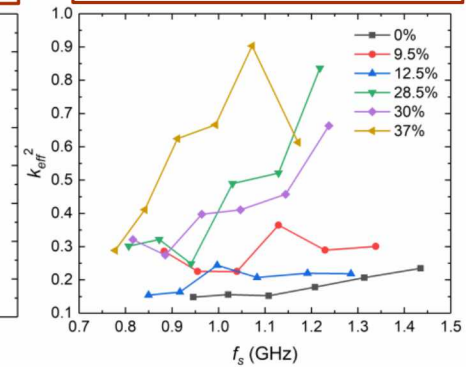


OUO

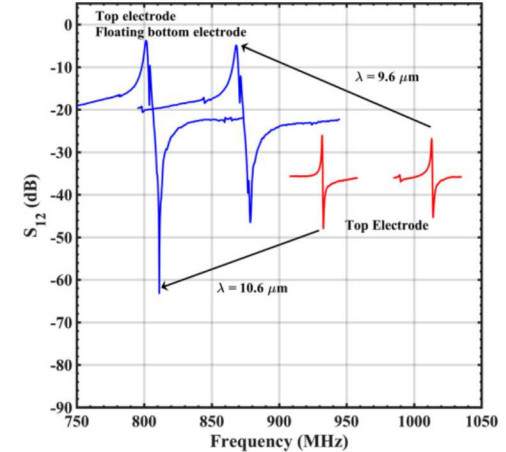
f_s decreases due to higher Sc alloyed films having lower E



Increase in electromechanical coupling with increasing Sc %

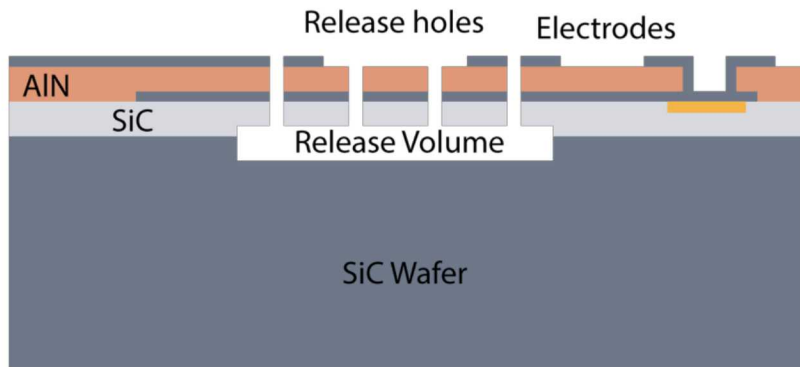


Henry et al., J. Vac. Sci. 36 (2018)





XMEMS Process



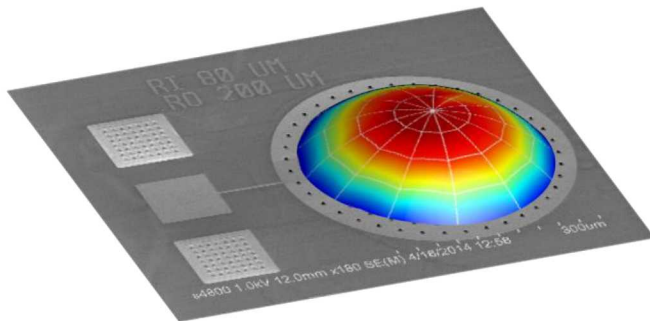
XMEMS process can fabricate devices capable of operating at elevated temperature

- AlN temperature limit $> 1000^{\circ}\text{C}$
- SiC temperature limit $> 2000^{\circ}\text{C}$

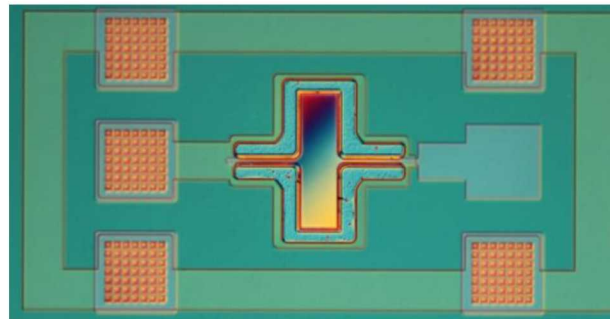
Devices fabricated

- **Ultrasonic transducer**
- **Microresonators**
- Accelerometers
- Phononic crystals
- Tuning forks

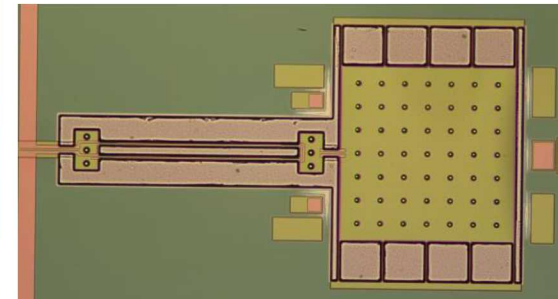
Piezoelectric Micromachined Ultrasonic Transducer (PMUT)



Microresonator

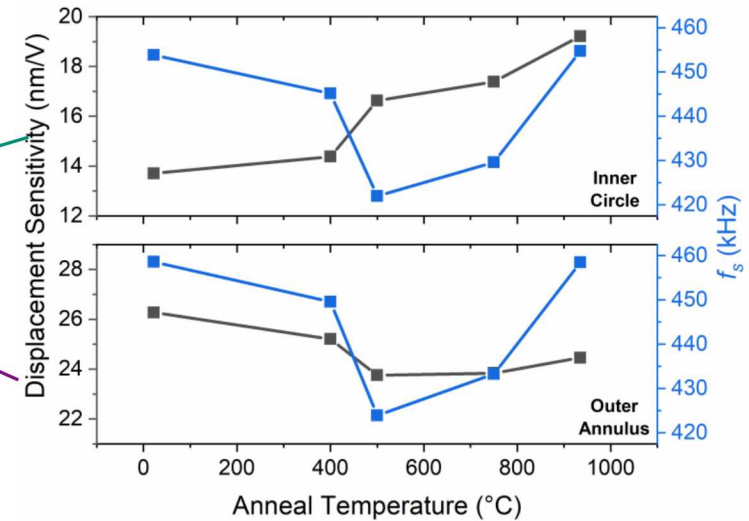
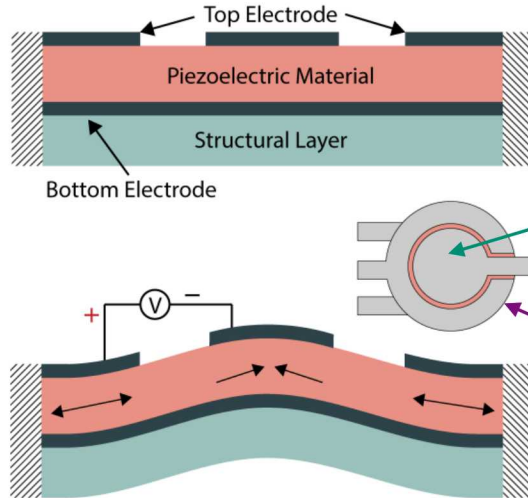


Accelerometer

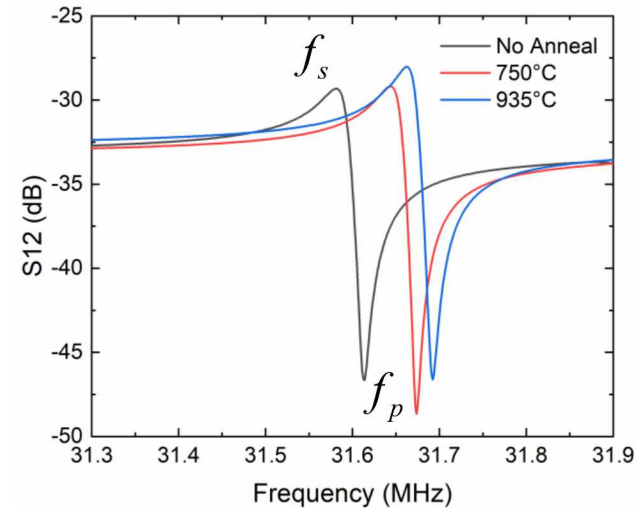
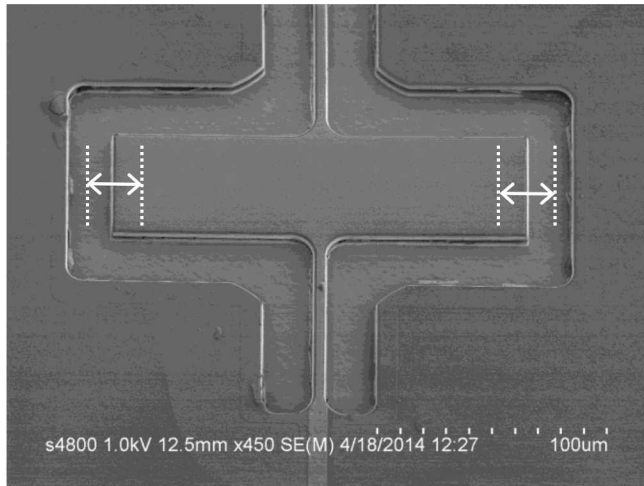




PMUT



Microresonator



Quality Factor

$$Q = \frac{f_s}{\Delta f_{3dB}}$$

Electromechanical Coupling

$$k_{eff}^2 = \frac{f_s^2 - f_p^2}{f_p^2}$$

Heated chuck from to 500°C in steps of 50°C

Temperature hysteresis loop results in permanent frequency shift in unannealed sample

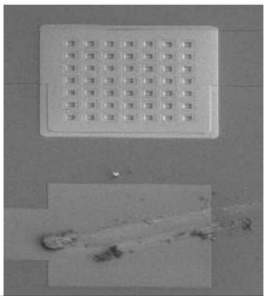
No degradation in piezoelectric performance

Challenges

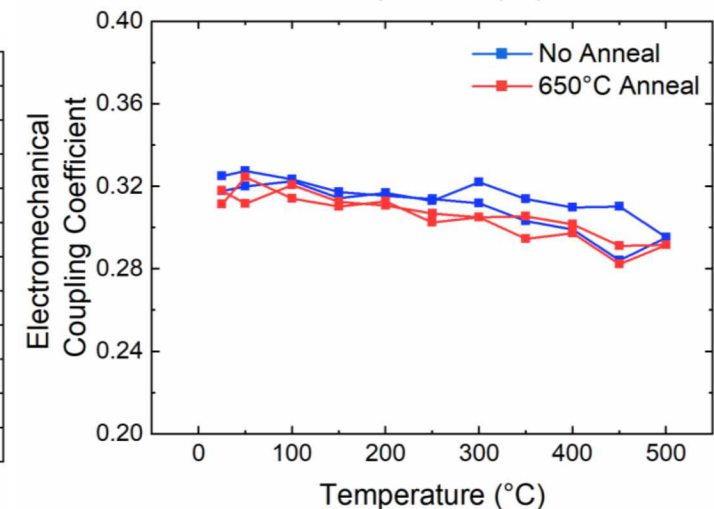
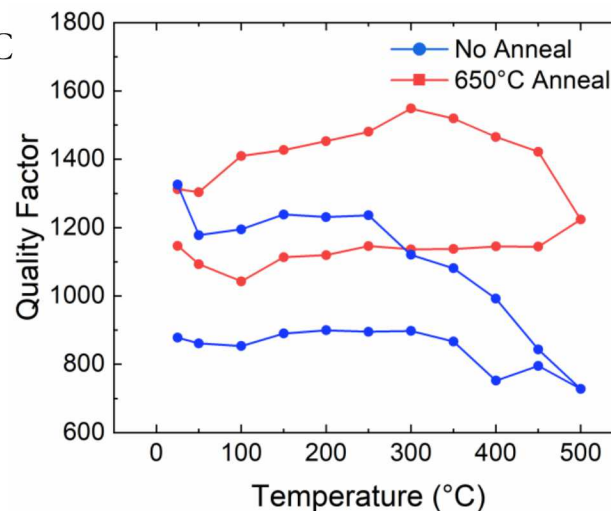
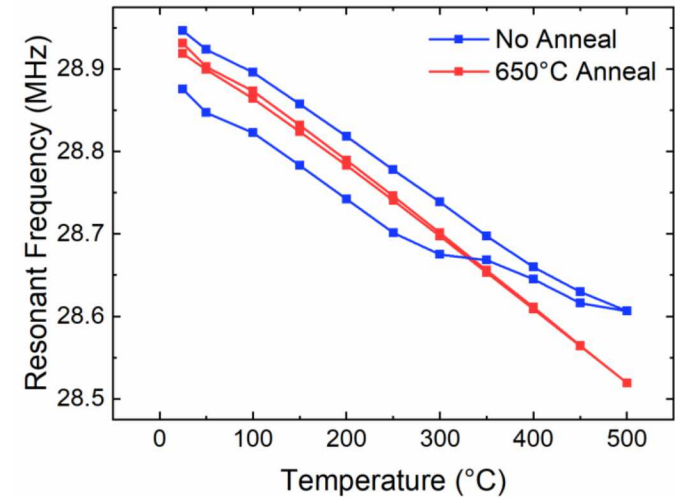
Quality factor degradation observed

Potential sources

- Probe contact issues
- Carbon contamination
- Via degradation
expected to form at 500°C



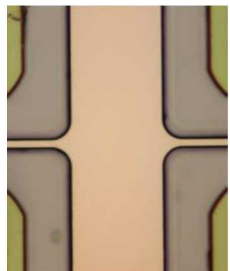
Probe tip shift
from thermal
expansion



- Heated chuck from to 500°C in steps of 50°C
- Temperature hysteresis loop results in permanent frequency shift in unannealed sample
- No degradation in piezoelectric performance

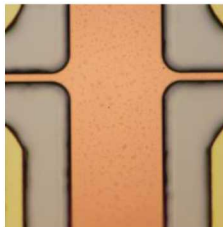
Challenges

- Quality factor degradation observed
 - Potential sources
 - Probe contact issues
 - Carbon contamination
 - Via degradation
 - TiO_2 is expected to form at 500 °C
- expected to form at 500°C

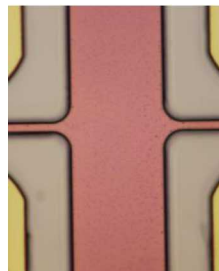


Post-release

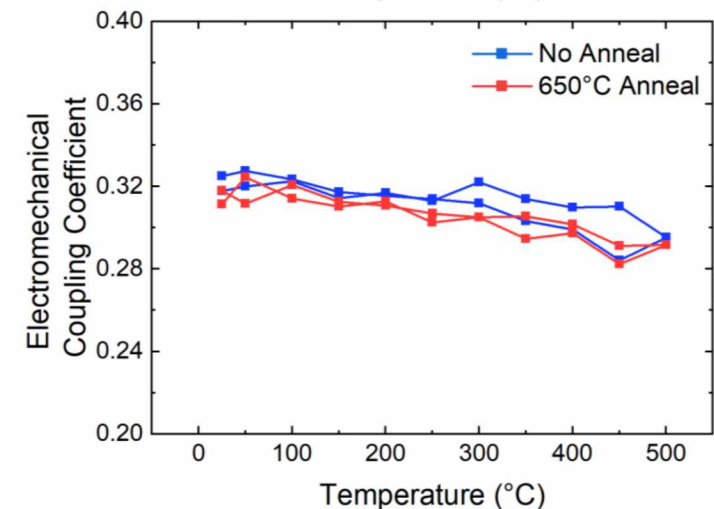
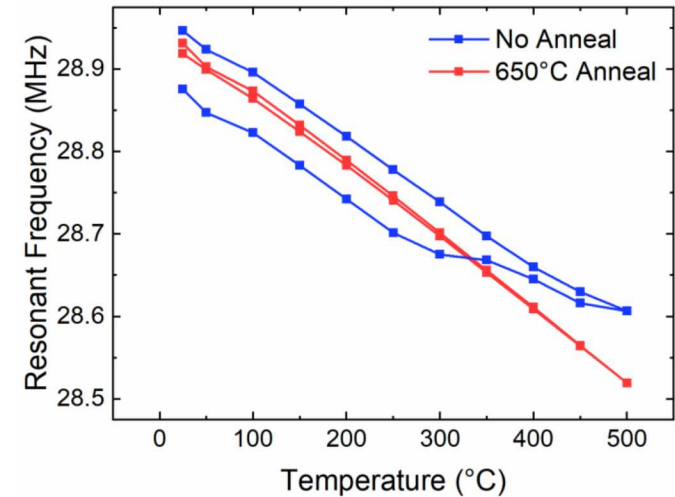
TiN Surface



Post-Anneal



Post-Air Test





Ti/Ti/AlCu_{0.05} allows for growth of Sc_{12.5}Al_{87.5}N films with a high-quality microstructure and texture

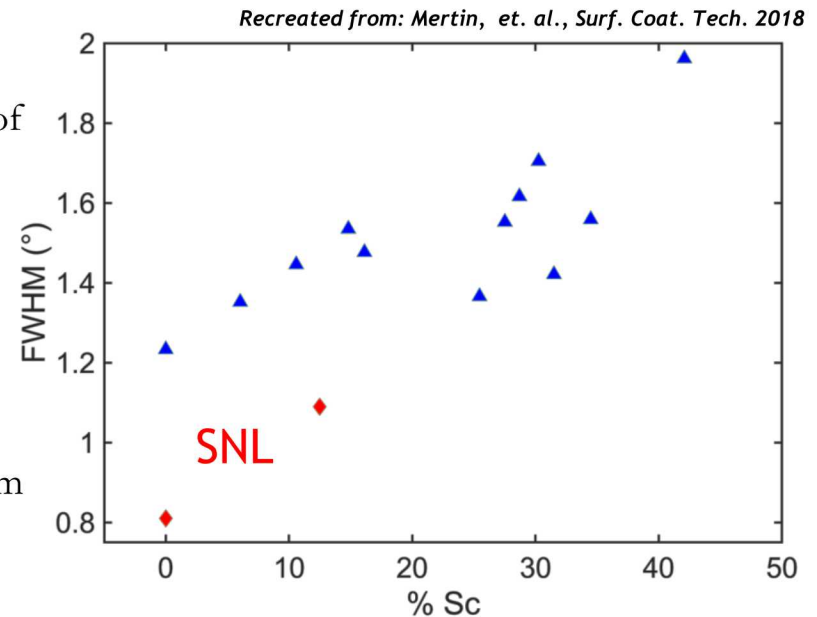
- Inclusion-free ScAlN
- CMOS compatible metal and allows for commercialization of ScAlN technologies

TiN is not needed to grow highly oriented AlN

- Can have significant impacts on device design

Increasing the thickness of Ti produces a well oriented AlN film

- Potential bottom metal for high temperature AlN technologies



Acknowledgements

- The authors acknowledge and thank the staff of Sandia's MESA fabrication facility for their help in depositing the metal stacks used in this study
- This project was supported by the LDRD program at Sandia National Laboratories