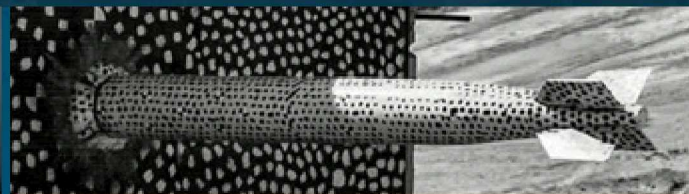


# Higher Voltage Iron Bipyridines for Non-Aqueous Redox Flow Batteries



PRESENTED BY

Claudina X. Cammack, Harry D. Pratt III, Leo J. Small, and Travis M. Anderson

## 2 Non-aqueous Flow Batteries

**Project Goal:** Build a better flow battery\* by targeting (1) Energy Density (2) Materials Cost (3) Mechanisms of Capacity Fade

$$\text{Energy Density}_{\text{RFB}} = \frac{1}{2} n F V_{\text{cell}} c_{\text{active}}$$

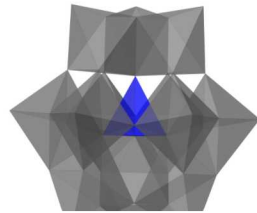
$$\text{ED}_{\text{AQ}} = \frac{1}{2} |F| 1.5 c_{\text{cell}} 2_{\text{active}} = 1.5F$$

$$\text{ED}_{\text{MetL}} = \frac{1}{2} 2F 2 c_{\text{cell}} 3_{\text{active}} = 6.0F$$

Low Cost Materials

Viscosity ☠☠

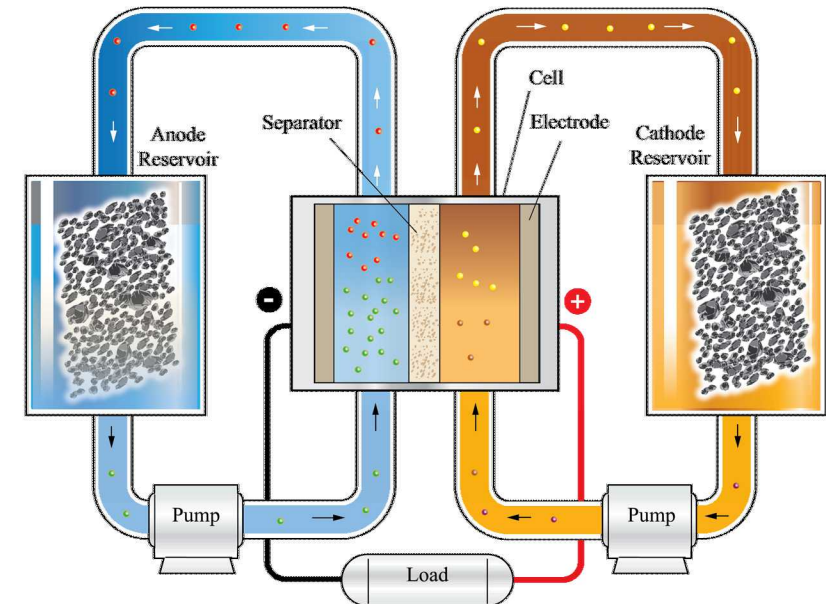
$$\text{ED}_{\text{MacM}} = \frac{1}{2} 2F 2 c_{\text{cell}} 0.1_{\text{active}} = 0.2F$$



$$\text{ED}_{\text{RedTarg}} = \frac{1}{2} |F| 3 c_{\text{cell}} 5_{\text{active}} = 7.5F$$

High Energy Density

Surface Area Kinetics ☠

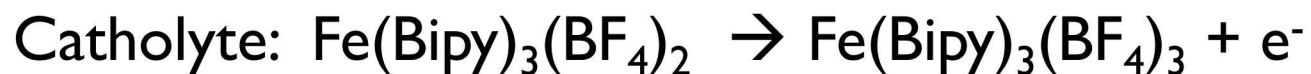


\* >4.4 mol e<sup>-</sup>/L according to Darling *et al.*, *Energy Environ. Sci.*, 2014, 7, 3459-3477.

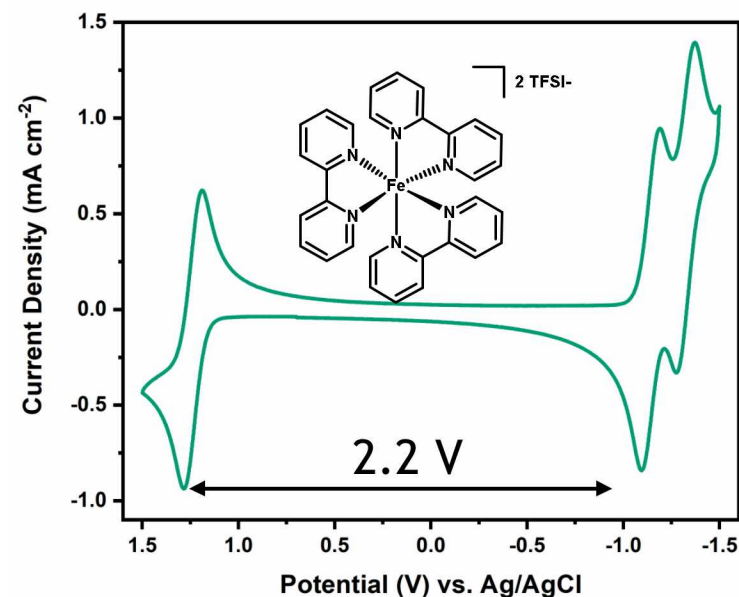
# Non-aqueous Flow Batteries-Metal Coordination Complexes (MCCs)

$$ED_{MCC} = \frac{1}{2} 2F 2.2_{\text{cell}} 0.2_{\text{active}} = 0.4F \quad \text{Low Cost Materials} \quad \text{Crossover} \quad \text{☠}$$

**First Generation** redox reactions (2.2 V)



**Second Generation** all-iron battery  
minimizes issues with crossover and  
utilizes *non-innocent* ligands.



SAMSUNG

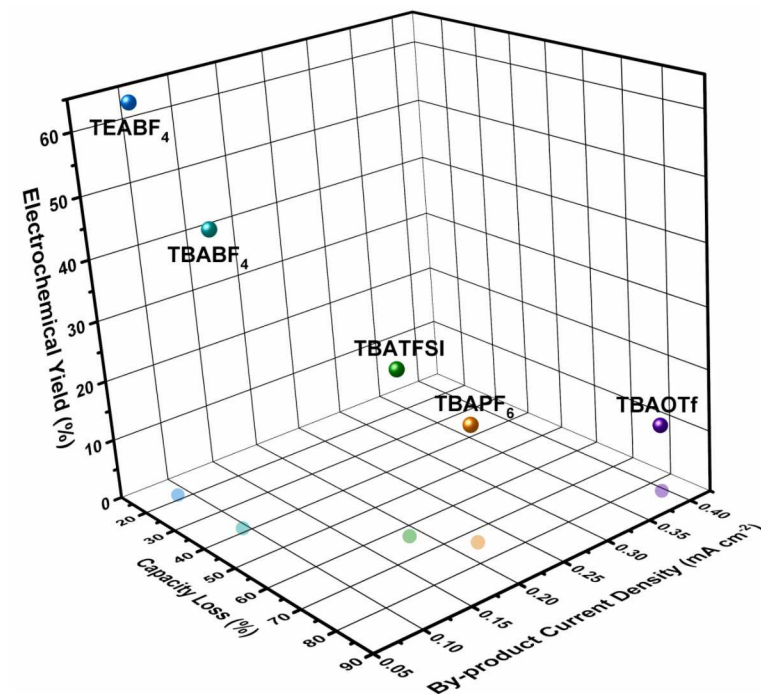
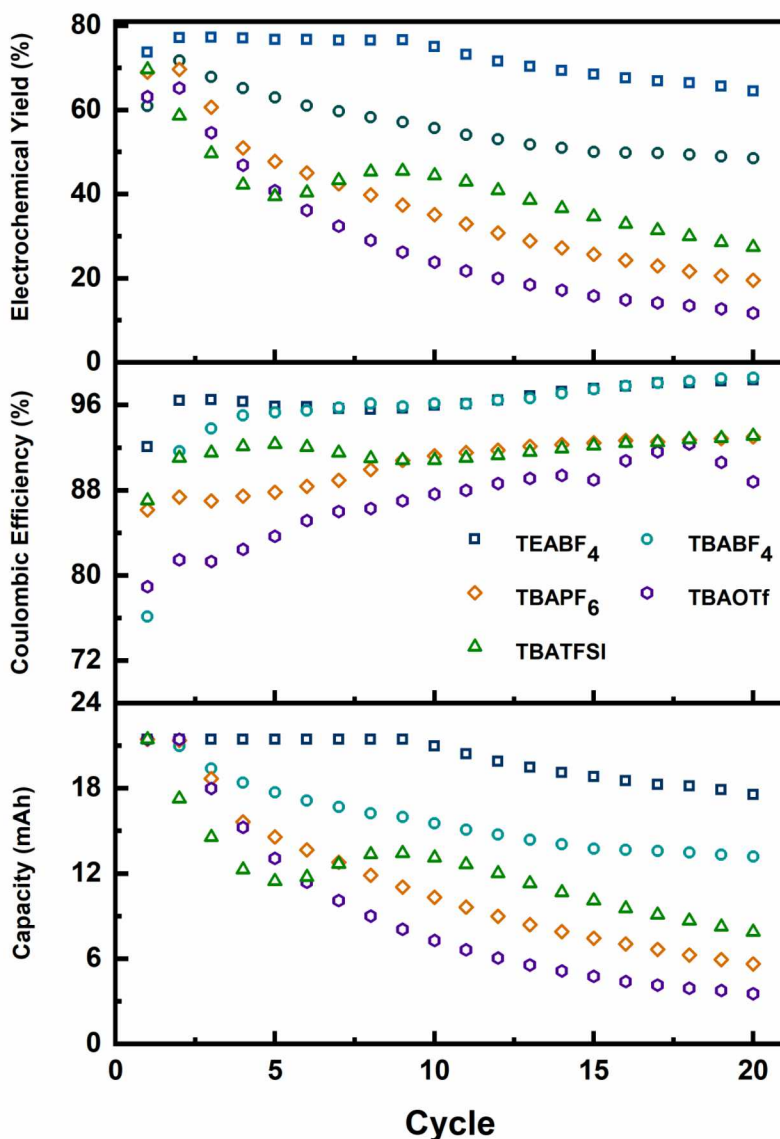
**Next Generation**

$$ED_{MCC} = \frac{1}{2} 2F 2.6_{\text{cell}} I_{\text{active}} = 2.6F$$

**Lower Symmetry, Higher Solubility**  
**Tunable Ligands, Wider Voltage**

## 4 Second Generation Salt Study

- Argon Glovebox, 0.5 M electrolyte, 0.2 M MCC, PC, unsupported AEM, 10 mA cm<sup>-2</sup>
- TEA<sup>+</sup>** is superior to TBA<sup>+</sup>
- BF<sub>4</sub><sup>-</sup>** is a superior anion
- TEABF<sub>4</sub>: **96.5 % CE**
- A by-product forms upon cycling
- Common decomposition mechanisms: ligand shedding and oxidative degradation
- Built RFBs with impure Fe(bpy)<sub>3</sub>(BF<sub>4</sub>)<sub>2</sub>
- 30 % impurity = 30% decrease in EY, CE, capacity
- Currently in the process of characterizing impurity

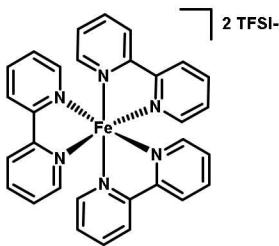


By-product formation  
correlates with RFB EY, CE,  
capacity loss

# 5 Tuning Bipyridine Ligands

Goal: produce a higher voltage symmetric RFB

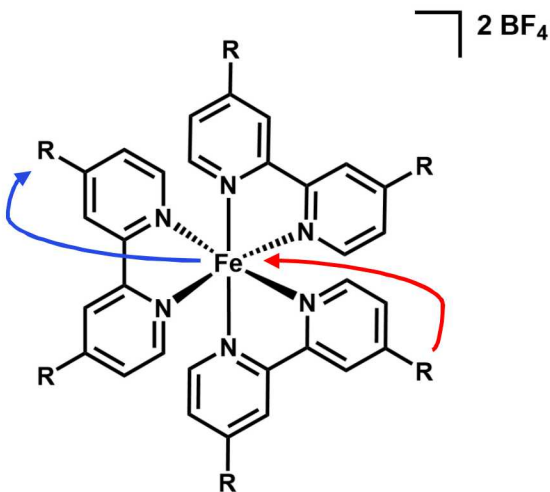
Samsung RFB



2.2 V

Fe(II)/(III)

Our complexes



R = CF<sub>3</sub>  
CO<sub>2</sub>Me  
Br  
H  
Me  
<sup>t</sup>Bu  
OMe  
NH<sub>2</sub>

Movement of e- density  
with substituents

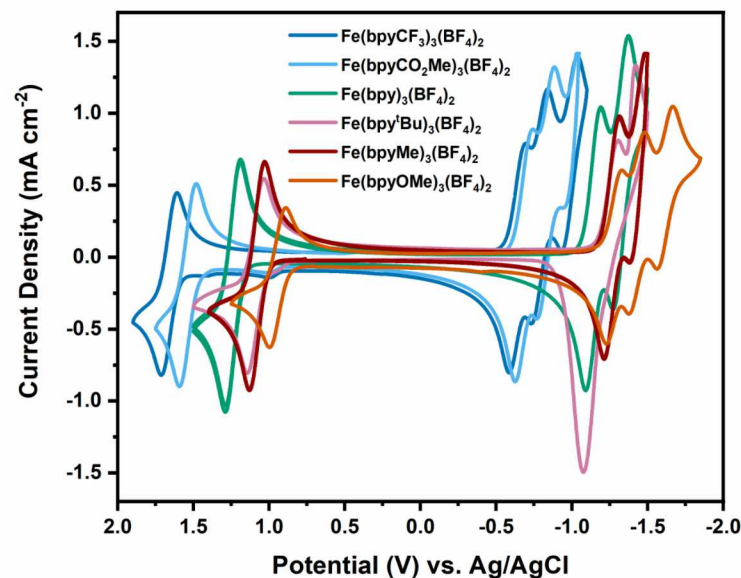
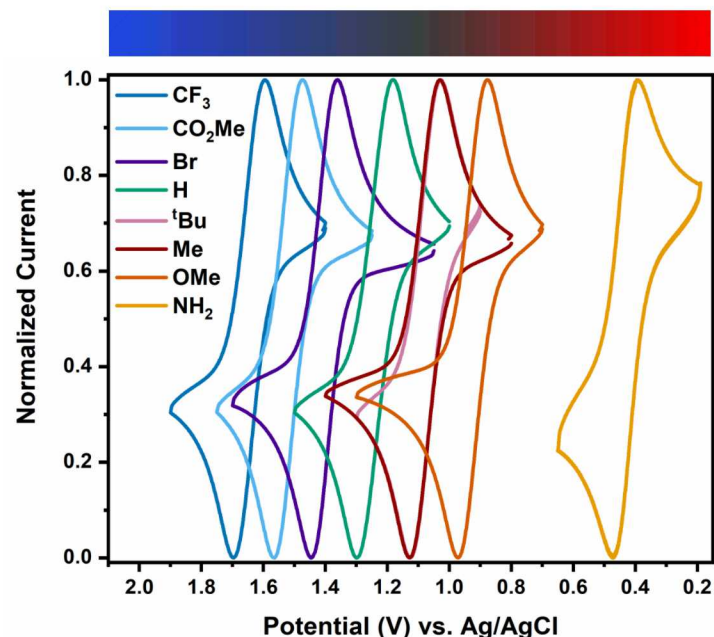
Substituent inductive effects

Electron-withdrawing  
More positive E<sub>1/2</sub>  
More difficult to oxidize Fe(II)

Electron-donating  
More negative E<sub>1/2</sub>  
Easier to oxidize Fe(II)

bipyridine: traditional π-acceptor ligand

# Ligand Effects on Redox Potentials



|                                                       | $\text{Fe}^{3+/2+} E_{1/2} \text{ (V)}$ | $\text{Fe}^{2+/+} E_{1/2} \text{ (V)}$ | $\Delta E_{1/2} \text{ (V)}$ |
|-------------------------------------------------------|-----------------------------------------|----------------------------------------|------------------------------|
| $\text{Fe}(\text{bpyCF}_3)_3(\text{BF}_4)_2$          | 1.65                                    | -0.63                                  | 2.28                         |
| $\text{Fe}(\text{bpyCO}_2\text{Me})_3(\text{BF}_4)_2$ | 1.53                                    | -0.68                                  | 2.21                         |
| $\text{Fe}(\text{bpyBr})_3(\text{BF}_4)_2$            | 1.43                                    | -                                      | -                            |
| $\text{Fe}(\text{bpy})_3(\text{BF}_4)_2$              | 1.25                                    | -1.12                                  | 2.37                         |
| $\text{Fe}(\text{bpy}^t\text{Bu})_3(\text{BF}_4)_2$   | 1.09                                    | -1.19                                  | 2.28                         |
| $\text{Fe}(\text{bpyMe})_3(\text{BF}_4)_2$            | 1.07                                    | -1.25                                  | 2.32                         |
| $\text{Fe}(\text{bpyOMe})_3(\text{BF}_4)_2$           | 0.94                                    | 1.27                                   | 2.21                         |
| $\text{Fe}(\text{bpyNH}_2)_3(\text{BF}_4)_2$          | 0.43                                    | -                                      | -                            |

$\underbrace{\hspace{1cm}}$   $\underbrace{\hspace{1cm}}$   $\underbrace{\hspace{1cm}}$   
 Fe(II)/(III)      ligand      voltage gap

Inductive effects change ease oxidation of Fe(II)

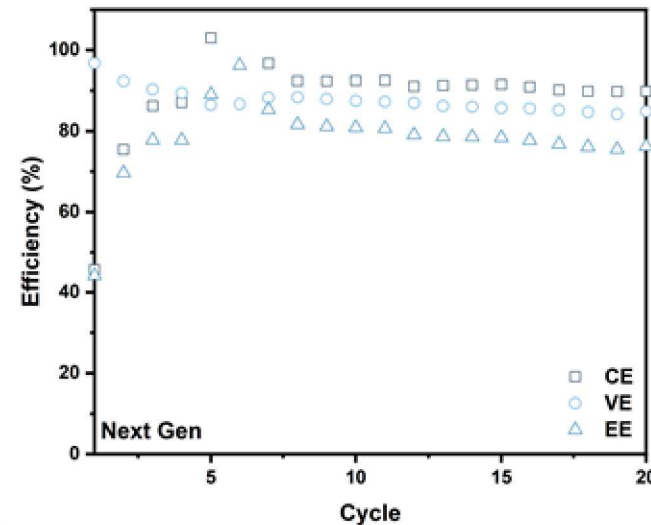
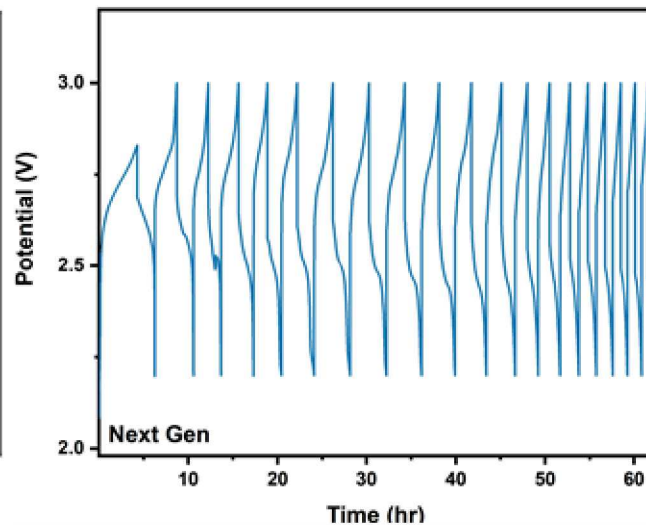
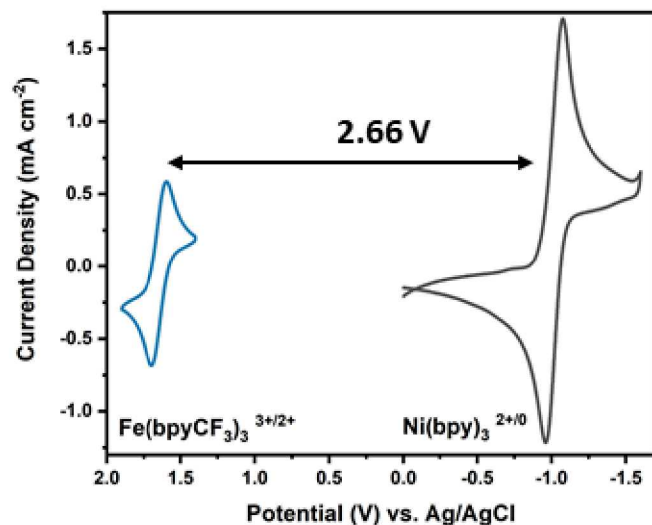
EWGs shifted **positively** by up to 0.4 V

EDGs shifted **negatively** by up to 0.8 V

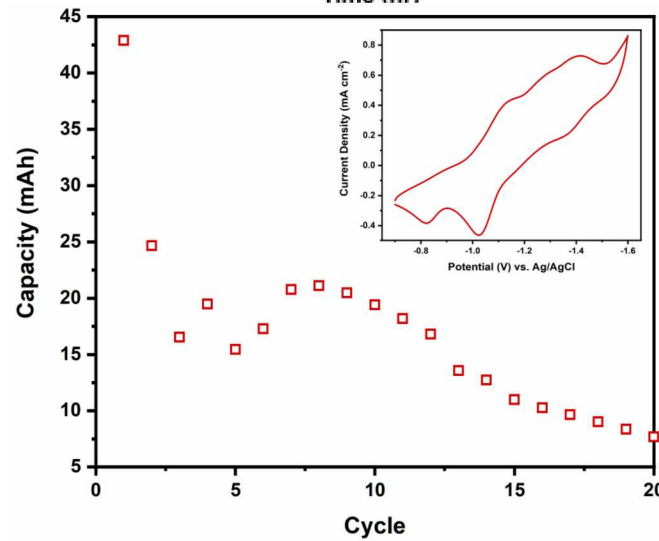
Fe(II) and ligand-centered redox shifted together

# 7 Asymmetric 2.6 V Fe-Ni RFB

$\text{Fe}(\text{bpyCF}_3)_3(\text{BF}_4)_2$  had highest catholyte  $E_{1/2}$  of 1.65 V (0.4 V increase)  
 Built  $\text{Fe}(\text{bpyCF}_3)_3(\text{BF}_4)_2/\text{Ni}(\text{bpy})_3(\text{BF}_4)_2$  to understand stability limits of cathode



Average CE and EE:  
**90.6 and 78.0 %** over 20  
 cycles

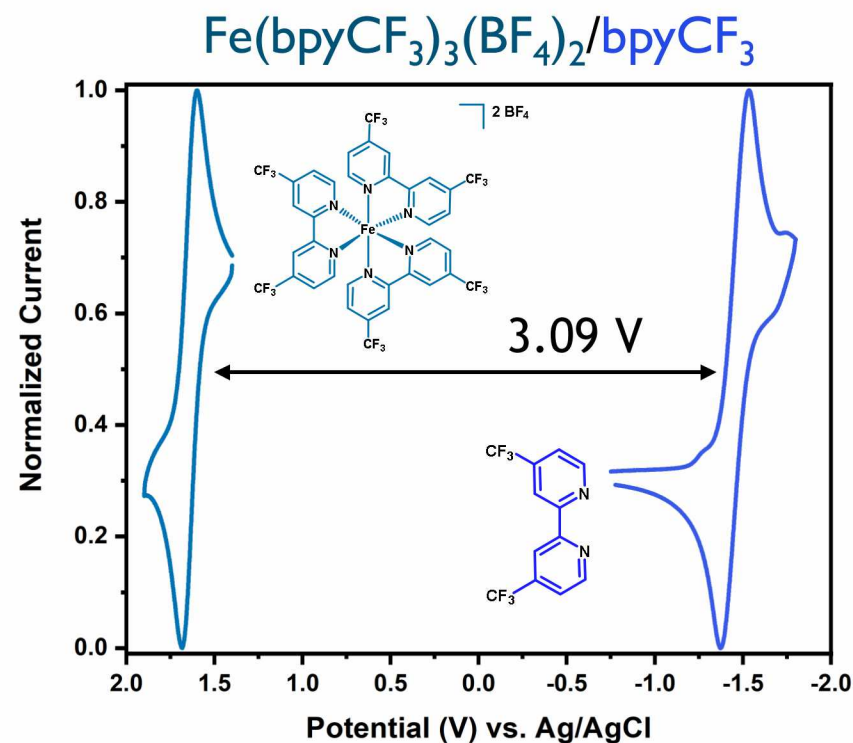
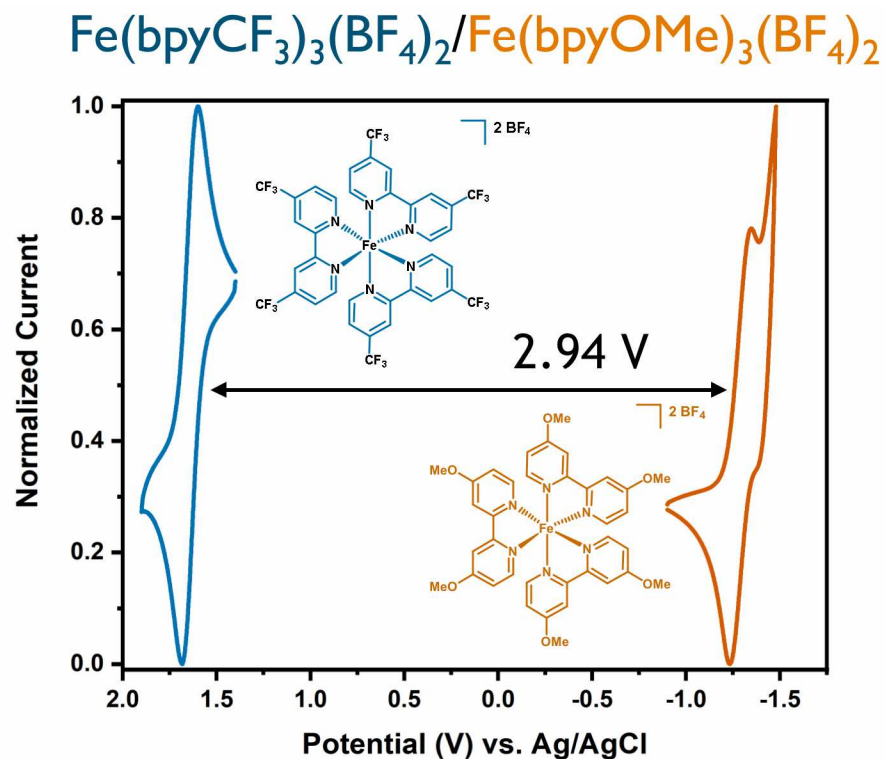


Catholyte showed cycling  
 stability at high voltage but  
 anolyte degraded, resulting in  
 capacity fade

We learned that:

- Crossover is less observed in symmetric RFBs
- $\text{Ni}(\text{bpy})_3(\text{BF}_4)_2$  has poor stability as an anolyte
- Symmetric  $\text{Fe}(\text{bpyR})_3(\text{BF}_4)_2$  redox couples shift together

*Pseudo*-symmetric RFBs may extend voltage window, show low crossover, and solve anolyte stability issue



## UNIVERSITY COLLABORATORS

Mitch Anstey

Davidson College

Ellen Matson

University of Rochester

Christopher Bejger

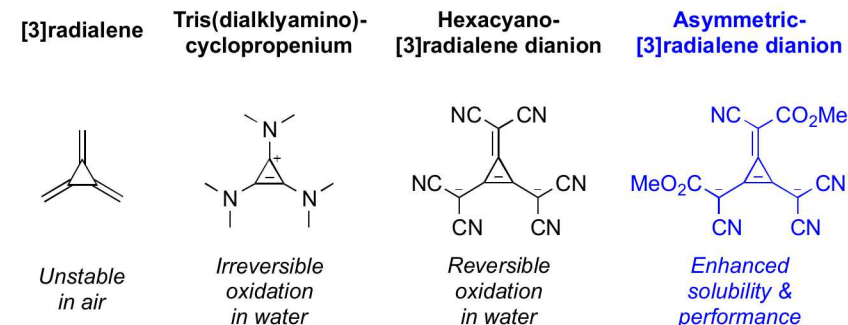
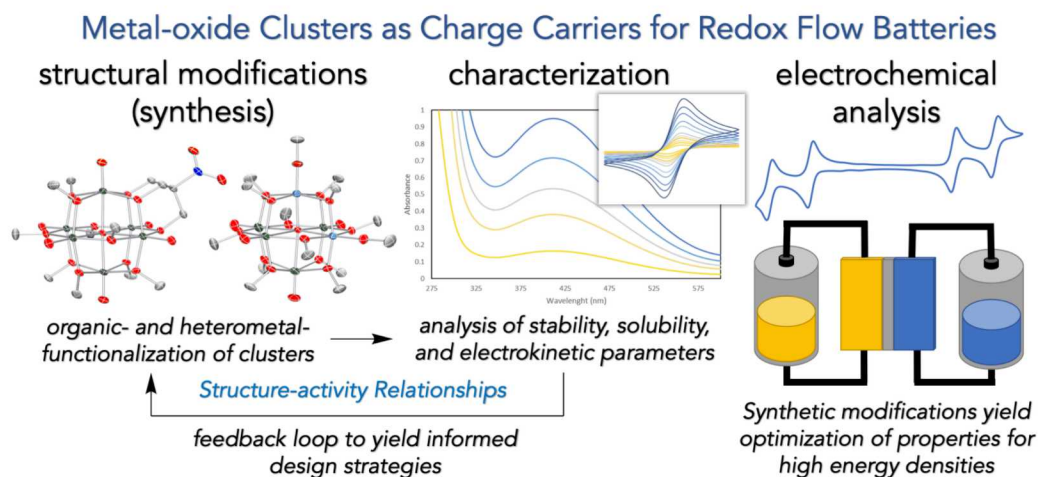
UNC Charlotte

## PROJECT

Redox molecules

Redox molecules

Redox molecules



## Technical Presentations

H. Pratt, L. Small, T. Anderson “Diagnostics for failure modes in non-aqueous flow batteries” 236<sup>th</sup> ECS Meeting, Atlanta, GA, October 13-17, 2019.

## Invited Talks

H. Pratt, L. Small, T. Anderson “Elucidating failure modes in non-aqueous flow batteries” 2020 Spring MRS Meeting and Exhibit, Boston, MA, November 28-December 4, 2020.

## Patents Issued

D. Sava Gallis, H. Pratt, T. Anderson, N. Hudak “Metal-Organic Framework Electrodes for Sodium Ion Batteries” U. S. Patent 10,497,971, December 3, 2019.

## Journal Publications

L. VanGelder, H. Pratt, T. Anderson, E. Matson “Surface functionalization of polyoxovanadium clusters: generation of highly soluble charge carriers for nonaqueous energy storage” Chemical Communications, October 18, 2019, vol. 55, 12247-12250,

<https://doi.org/10.1039/C9CC05380H>

N. Turner, M. Freeman, H. Pratt, A. Crockett, D. Jones, M. Anstey, T. Anderson, C. Bejger “Desymmetrized hexasubstituted [3]radialene anions as aqueous organic catholytes for redox flow batteries” Chemical Communications, March 4, 2020, vol. 56, 2739-2742,

<https://doi.org/10.1039/C9CC08547E>

THIS WORK WAS SUPPORTED THROUGH THE ENERGY STORAGE PROGRAM, MANAGED BY DR. IMRE GYUK, WITHIN THE U.S. DEPARTMENT OF ENERGY'S OFFICE OF ELECTRICITY.

Travis Anderson, PI

Sandia National Laboratories

[tmander@sandia.gov](mailto:tmander@sandia.gov)

505-844-4766