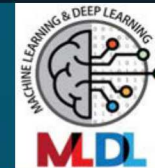


ACCELERATING PHASE-FIELD BASED PREDICTIONS VIA SURROGATE MODELS TRAINED BY MACHINE LEARNING METHODS

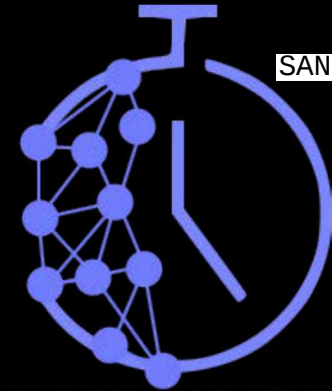
PRESENTED BY

RÉMI DINGREVILLE

DAVID MONTES DE OCA ZAPIAN & JAMES STEWART



2020 MLDL Workshop

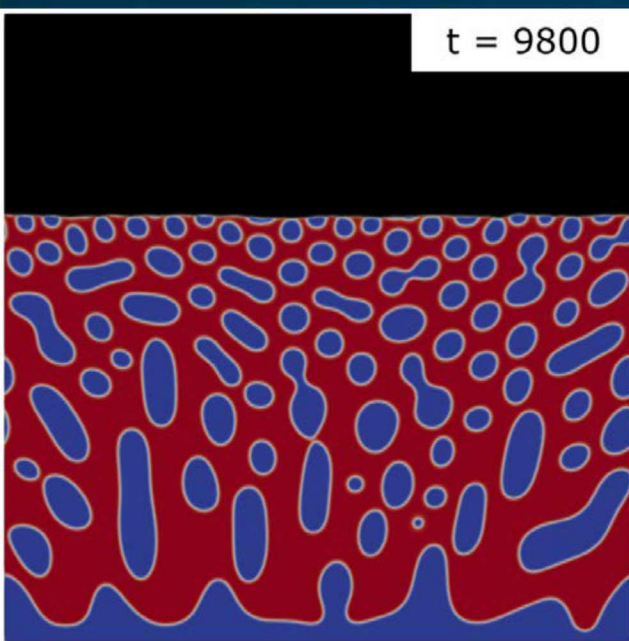


SAND2020-8314PE

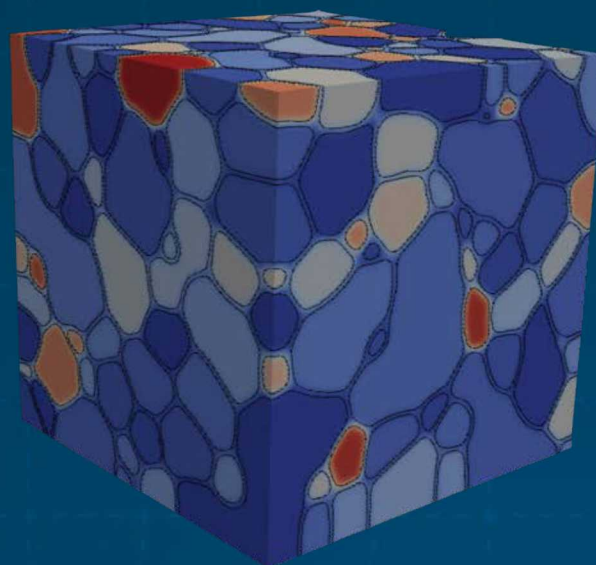
PREDICTING THE EVOLUTION OF MICROSTRUCTURES



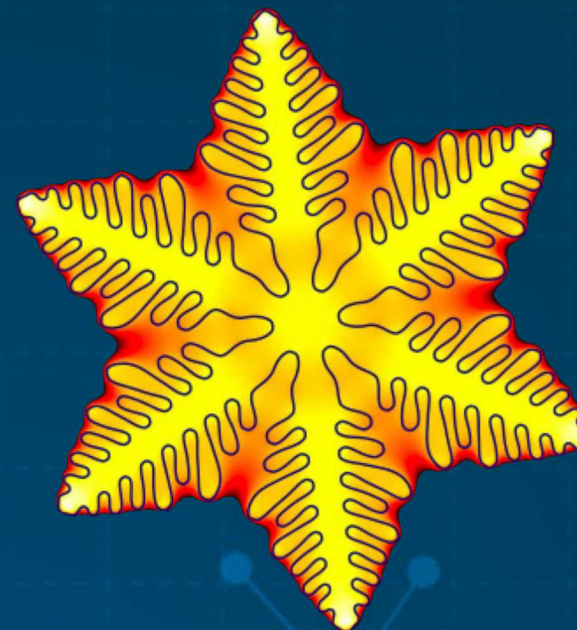
Thin-film deposition



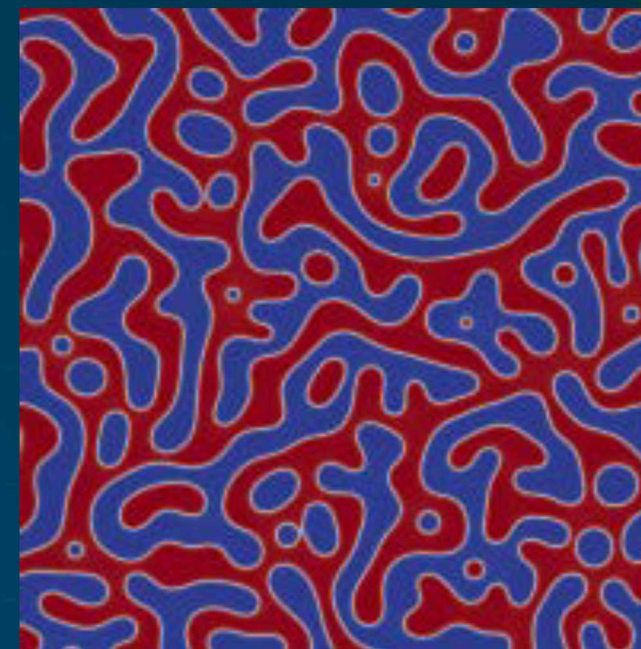
Grain growth



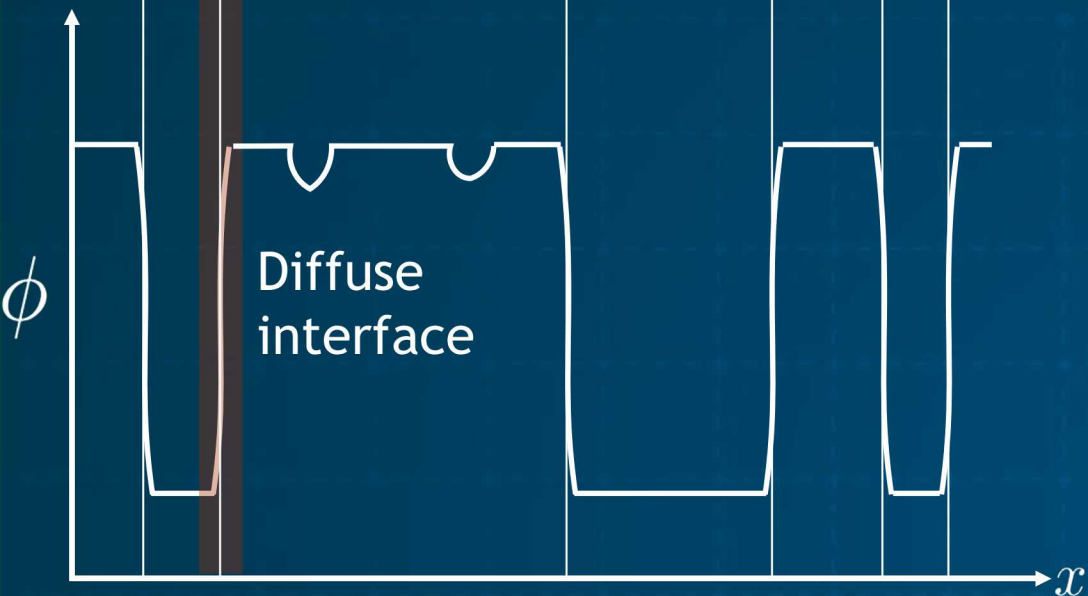
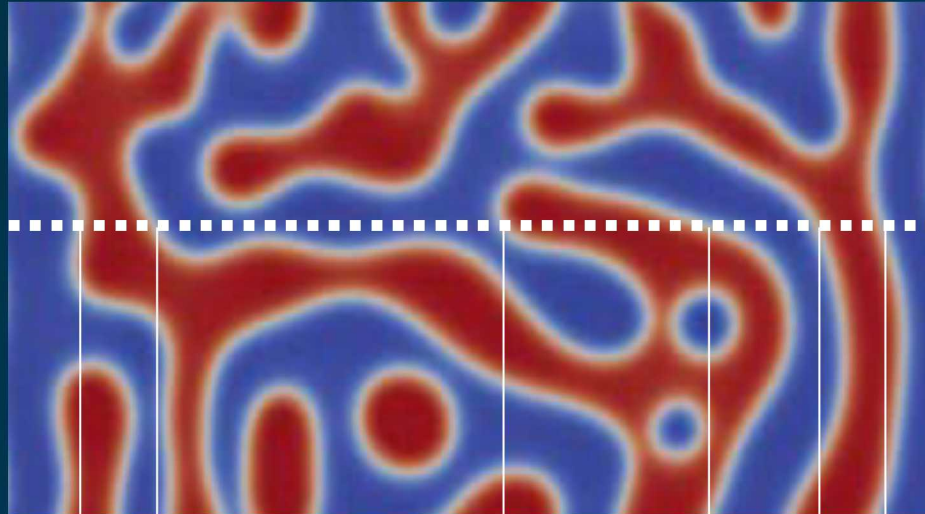
Solidification



Phase separation



PHASE FIELD 101



$$F = \int \left[f(\eta_1, \dots, \eta_p, c_1, \dots, c_p) + \sum_{i=1}^n \alpha_i (\nabla c_i)^2 + \sum_{i,j=1}^3 \sum_{k=1}^p \beta_{ij} \nabla_i \eta_k \nabla_j \eta_k \right] d\Omega$$

Allen-Cahn:

Non-conserved

$$\frac{\partial \eta_p}{\partial t} = -M_{pq} \frac{\delta F}{\delta \eta_q}$$

Cahn-Hillard:

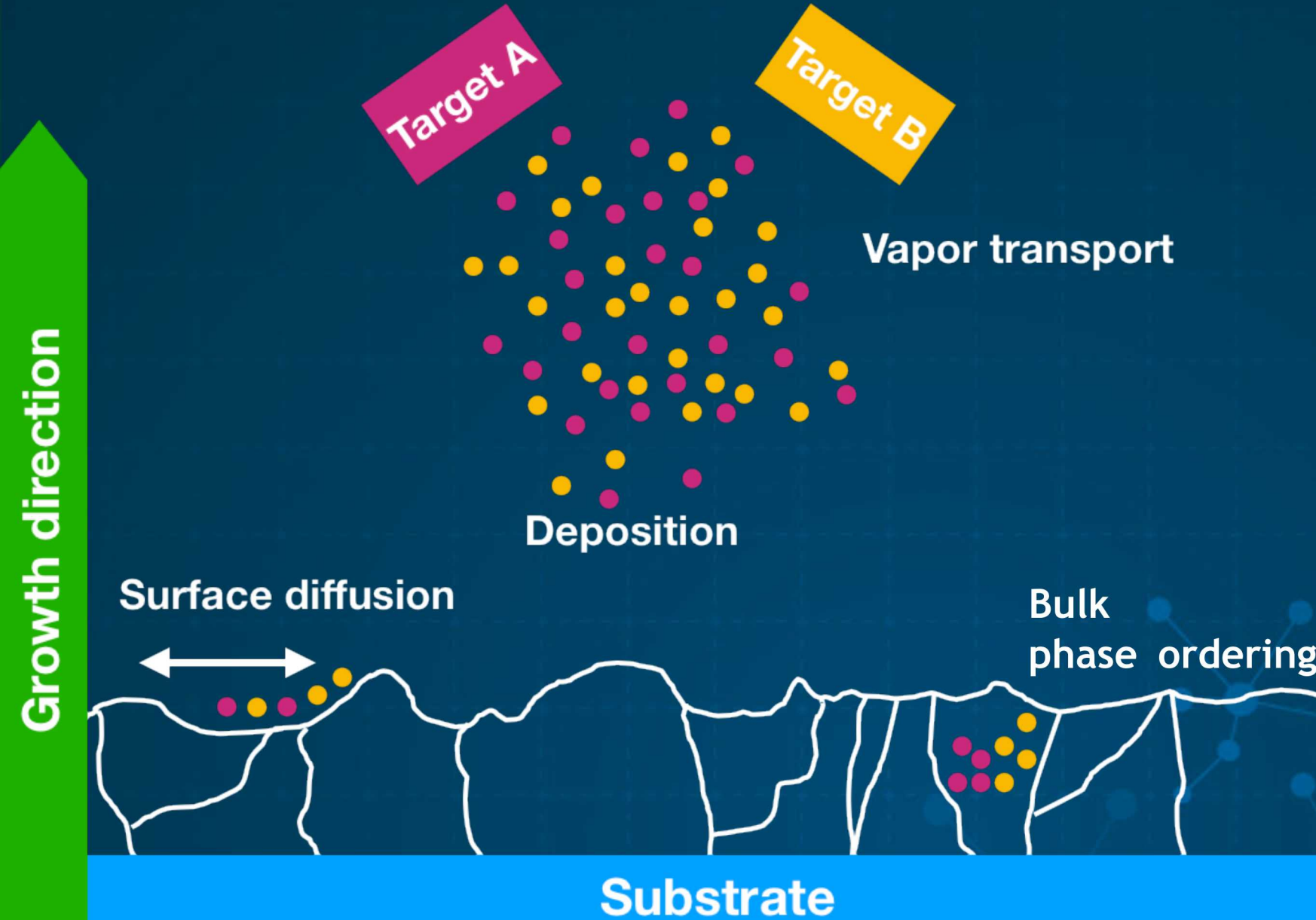
Conserved

$$\frac{\partial c_i}{\partial t} = \nabla \cdot \left(M_{ij} \nabla \frac{\delta F}{\delta c_j} \right)$$

EXAMPLE: PHYSICAL VAPOR DEPOSITION OF THIN FILMS



MODELING PVD USING PHASE-FIELD



1. **Explicitly model vapor transport**
...so we can model multiple PVD processes
2. **No artificial deposition of vapor**
...so we can capture hillock formation
3. **Account for surface and bulk diffusion into one model**
...so we can account for the composition of surface vs. bulk phenomena

MODELING PVD USING PHASE-FIELD

$$F = \int \left[\boxed{f_{\phi} + \frac{\kappa_{\phi}^2}{2} (\nabla \phi)^2} + s(\phi) \left(\boxed{f_{el} + f_c + \frac{\kappa_c^2}{2} (\nabla c)^2} \right) \right] d\Omega$$

Describing vapor/solid field

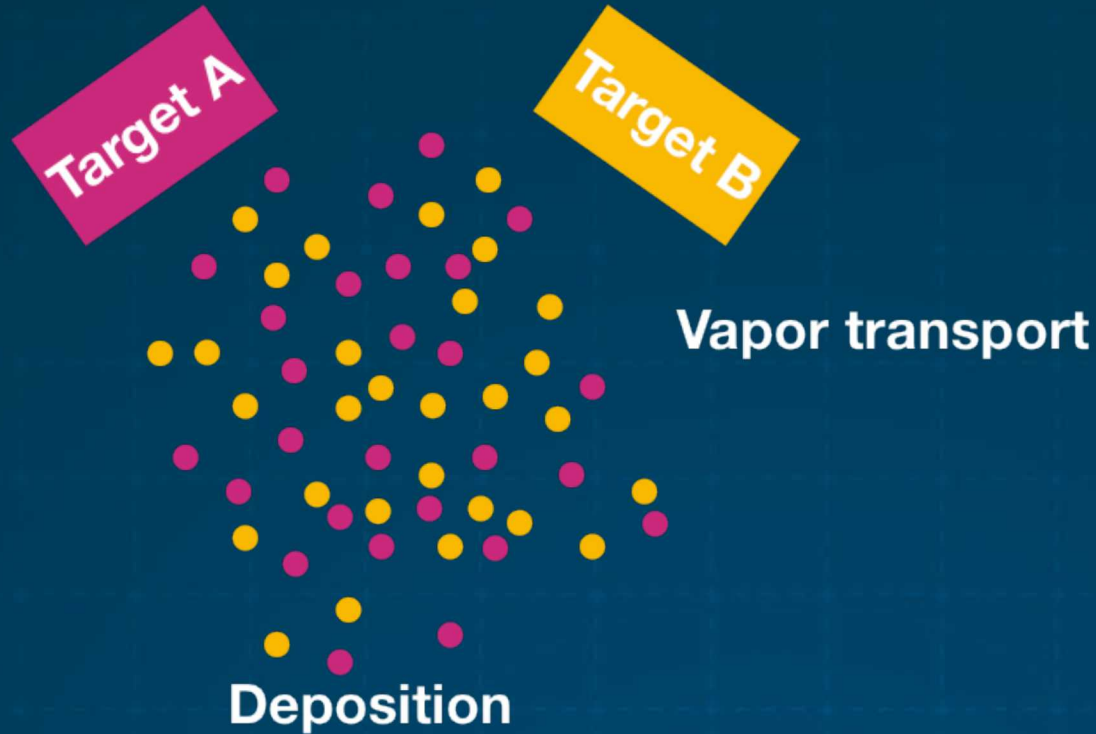
Only considered within the solid thin film

Describes concentration & elastic fields

Describes two-way coupling between elastic and concentration fields (Vegard)

Substrate

TRANSPORT & EVOLUTION OF INCIDENT VAPOR



1. Local vapor pressure
2. Sputtering power (velocity)
3. Vapor density and velocity can be adapted to simulated various PVD processes
4. Random distribution of vapor phase initially
5. Source term to allow for removal of vapor that is being converted to solid at thin-film interface
6. Inclusion of LB capability in the future to include hydrodynamic effects (ALD, CVD)

$$\frac{\partial \rho}{\partial t} = \nabla \cdot (D(\rho) \nabla \rho) - \nabla \cdot (\rho \vec{u}) - S(\vec{n})$$

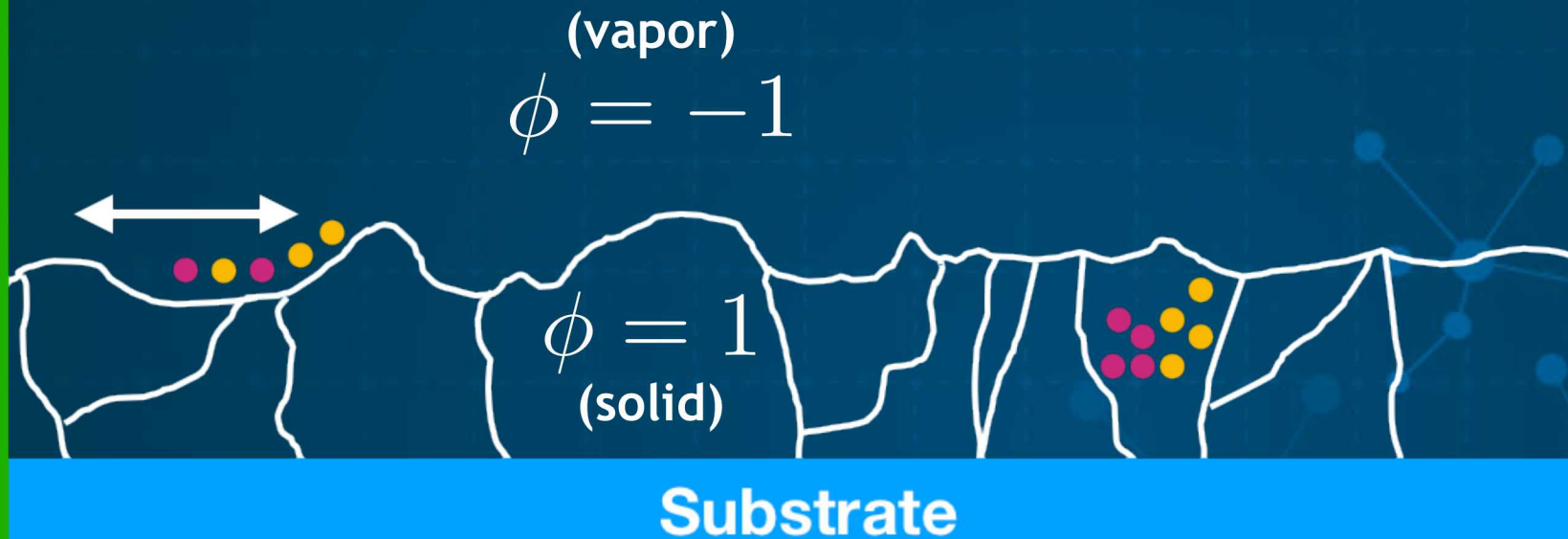
Convection-diffusion

Substrate

KINETICS OF THIN-FILM GROWTH

$$\frac{\partial \phi}{\partial t} = \nabla \cdot \left(M(\phi) \nabla \frac{\delta F}{\delta \phi} \right) + S(\vec{n})$$

1. Allows for arbitrary surface morphology formation
2. Captures surface-diffusion effects
3. Coupling of thin-film evolution to incident-vapor flux

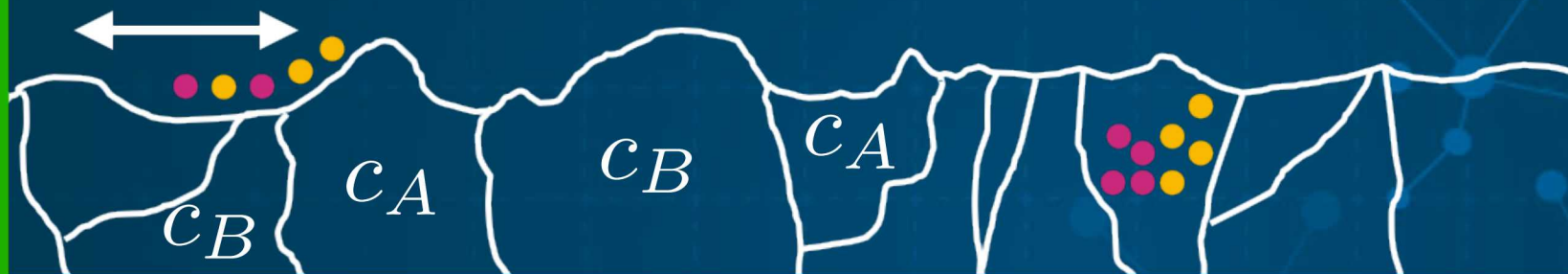


PHASE ORDERING IN THIN-FILM

$$\frac{\partial c}{\partial t} = \nabla \cdot \left(M(\phi, c) \nabla \frac{\delta F}{\delta c} \right)$$

$$M_c(\phi, c) = M^{\text{Bulk}} + M^{\text{Surf}}$$

1. Bulk or surface mobilities are selected through a ϕ -dependent switching function
2. Ensure that surface mobility is localized to vapor-solid interface
3. Explicitly capture both surface and bulk diffusion



Substrate

Growth direction

MODELING PVD USING PHASE-FIELD

Target A

Lateral

Target B

Target A

Vertical

Target B

Target A

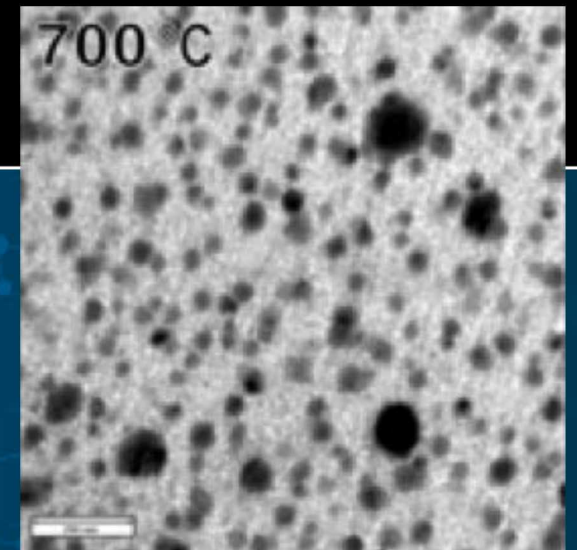
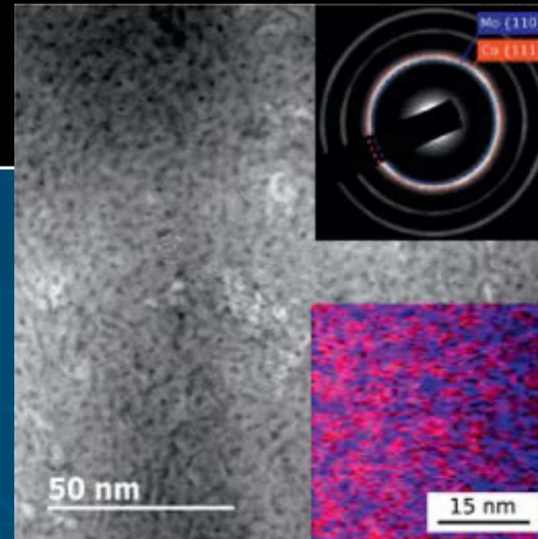
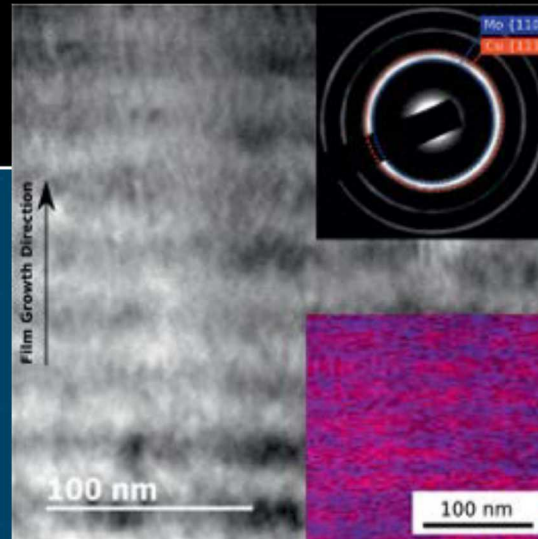
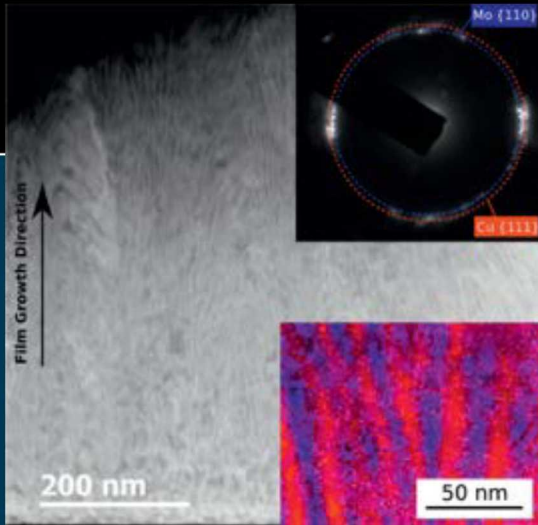
Random

Target B

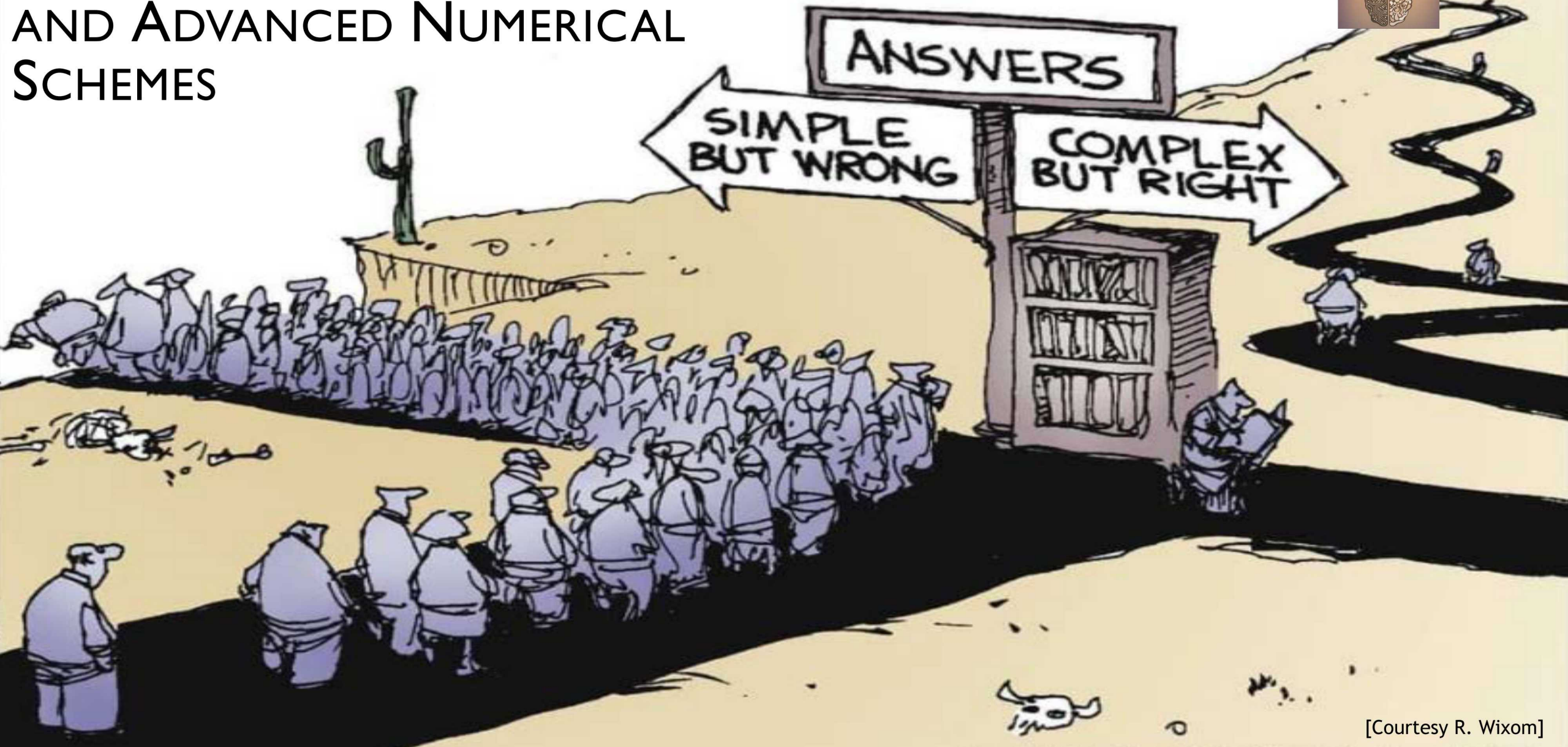
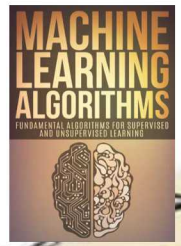
Target A

Nanoprecipitate

Target B

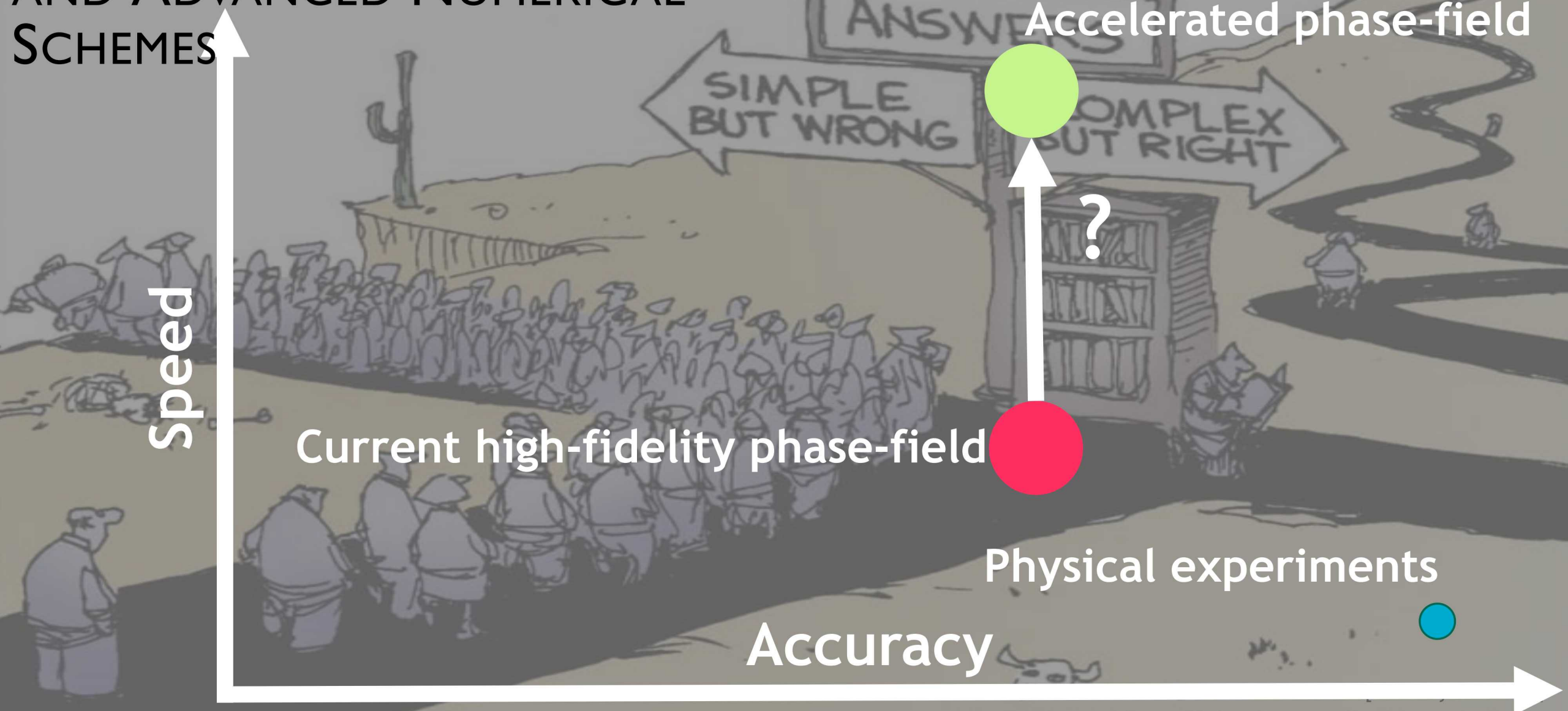
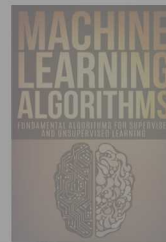


CURRENT EFFORTS FOCUS ON LEVERAGING HIGH-PERFORMANCE COMPUTING ARCHITECTURES AND ADVANCED NUMERICAL SCHEMES



[Courtesy R. Wixom]

CURRENT EFFORTS FOCUS ON LEVERAGING HIGH-PERFORMANCE COMPUTING ARCHITECTURES AND ADVANCED NUMERICAL SCHEMES

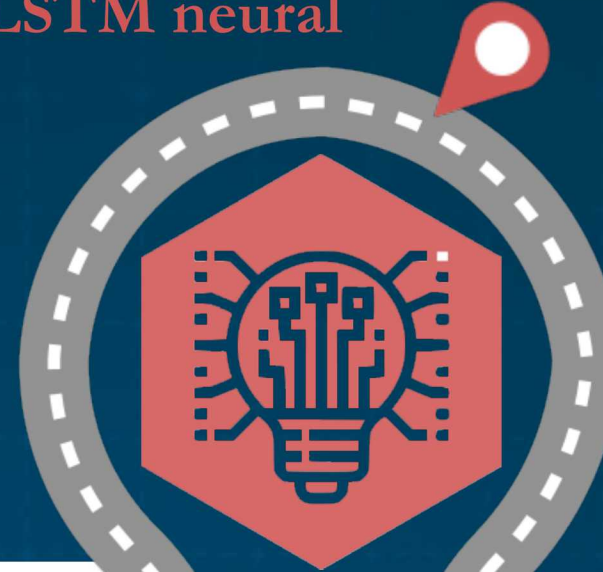


ALGORITHMIC APPROACH

1. Data preparation



3. LSTM neural



5. Accelerated PF



2. Low-Dimensional representation

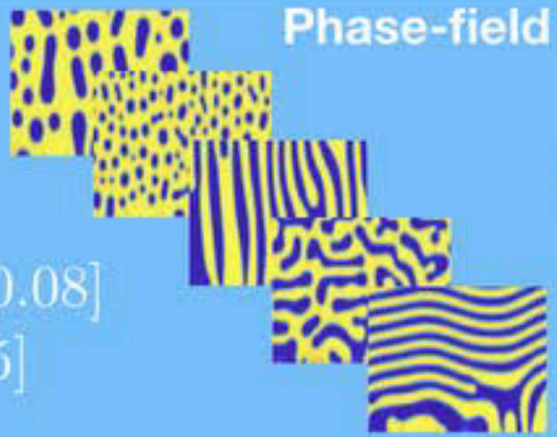


4. ML-trained surrogate

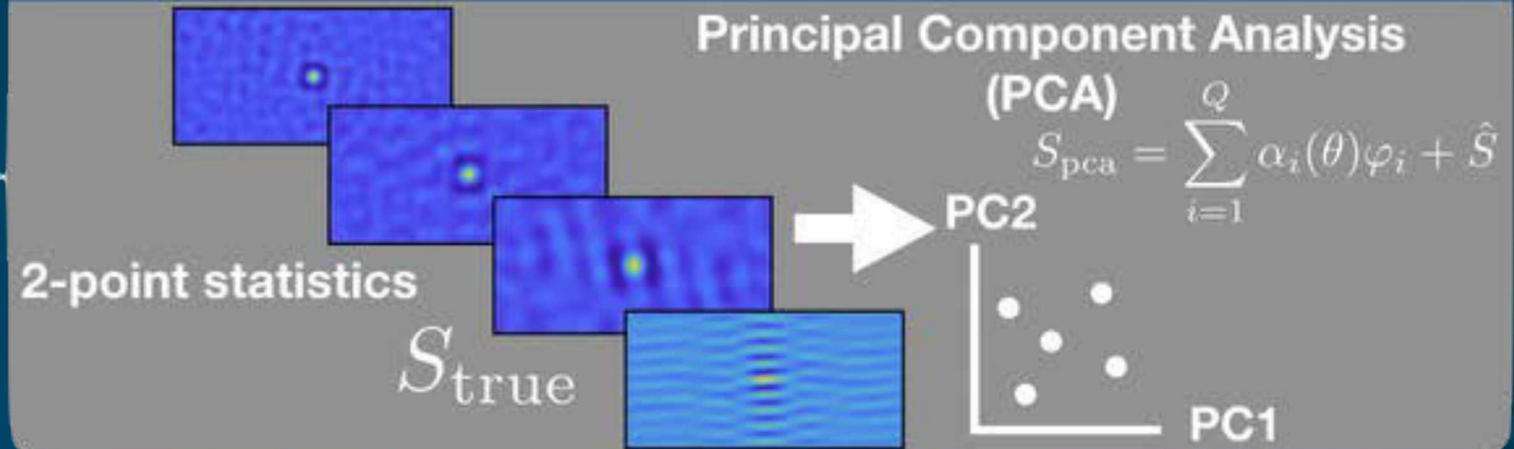


DIMENSIONALITY REDUCTION

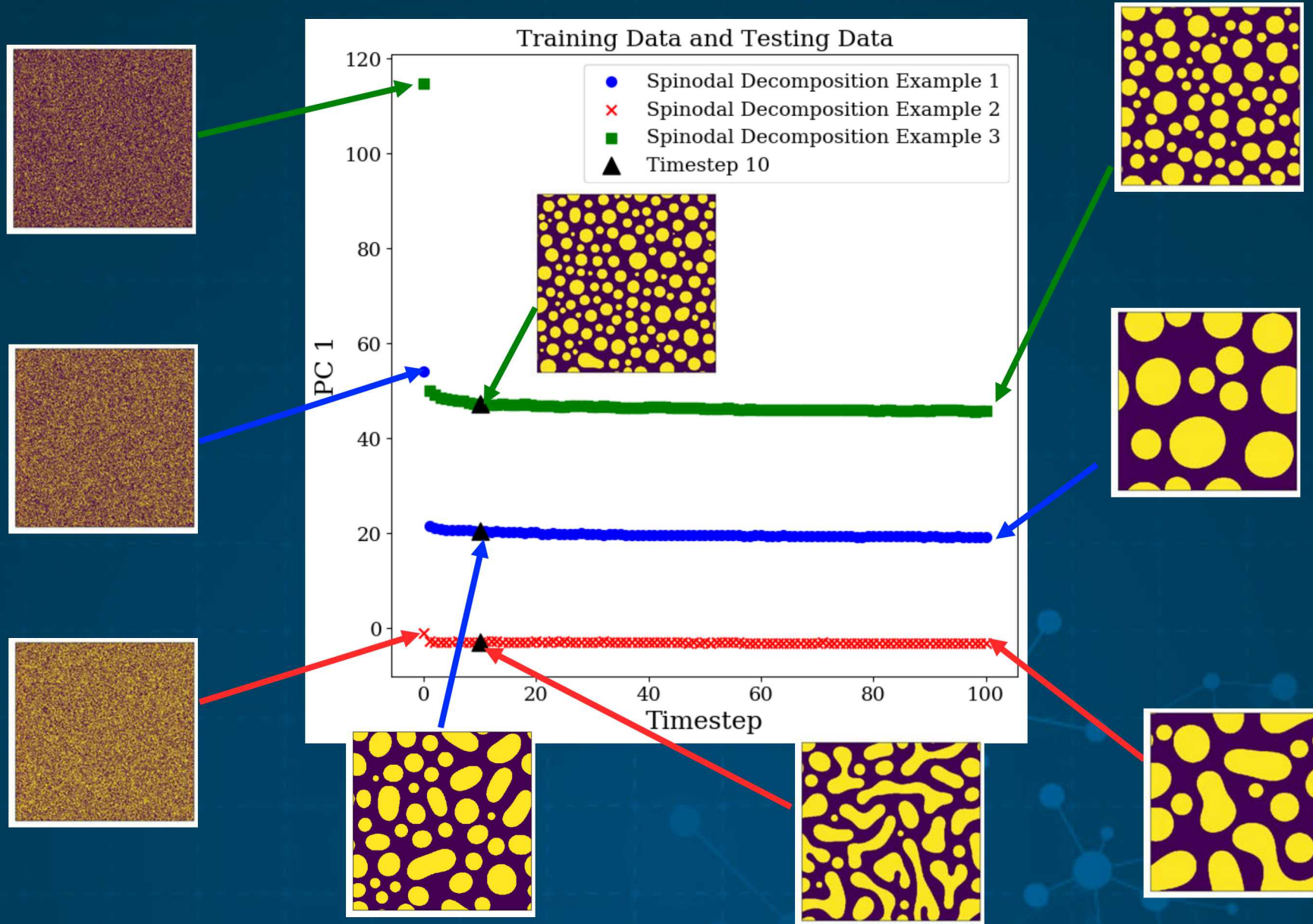
1. Simulation by phase-field



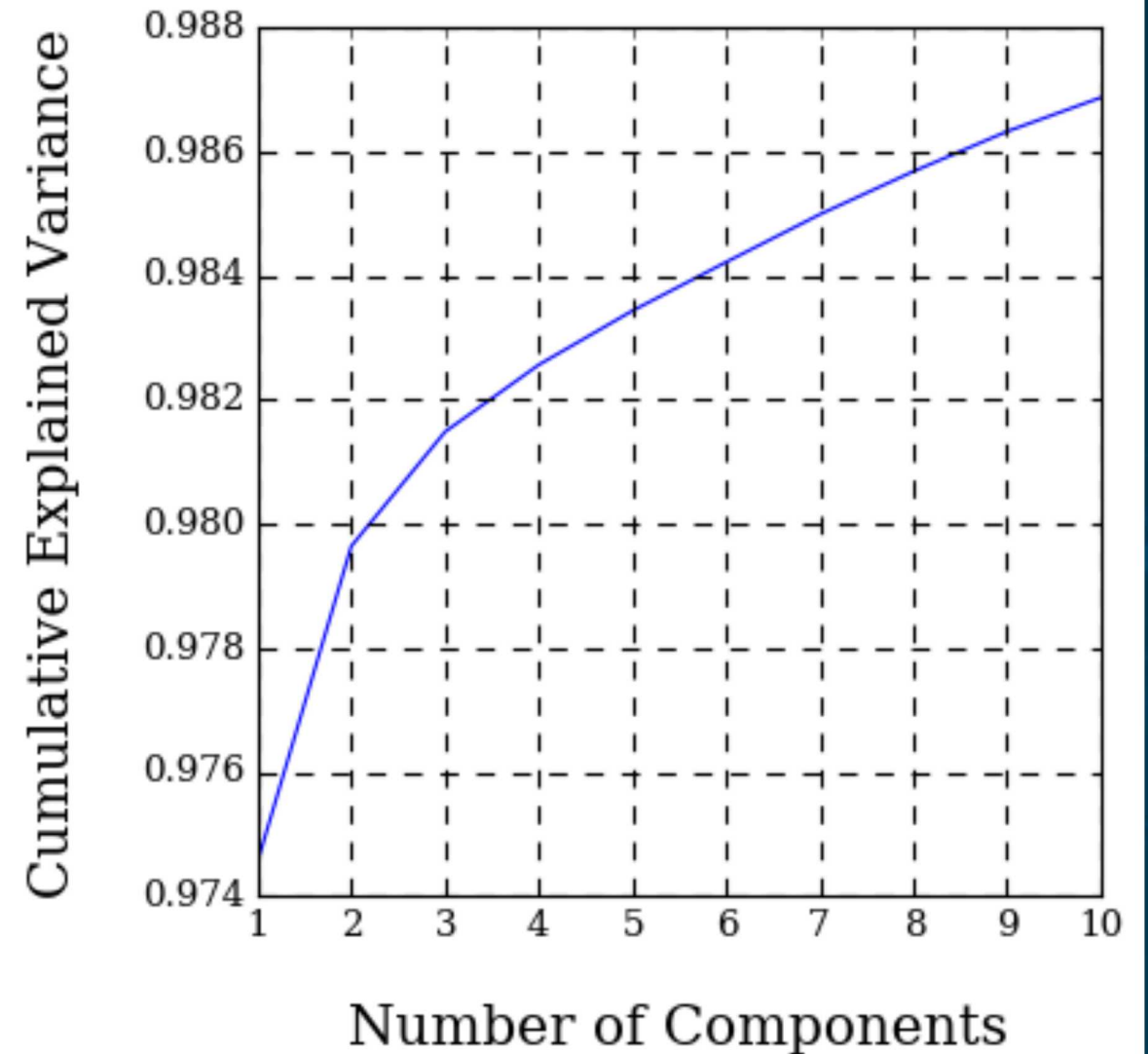
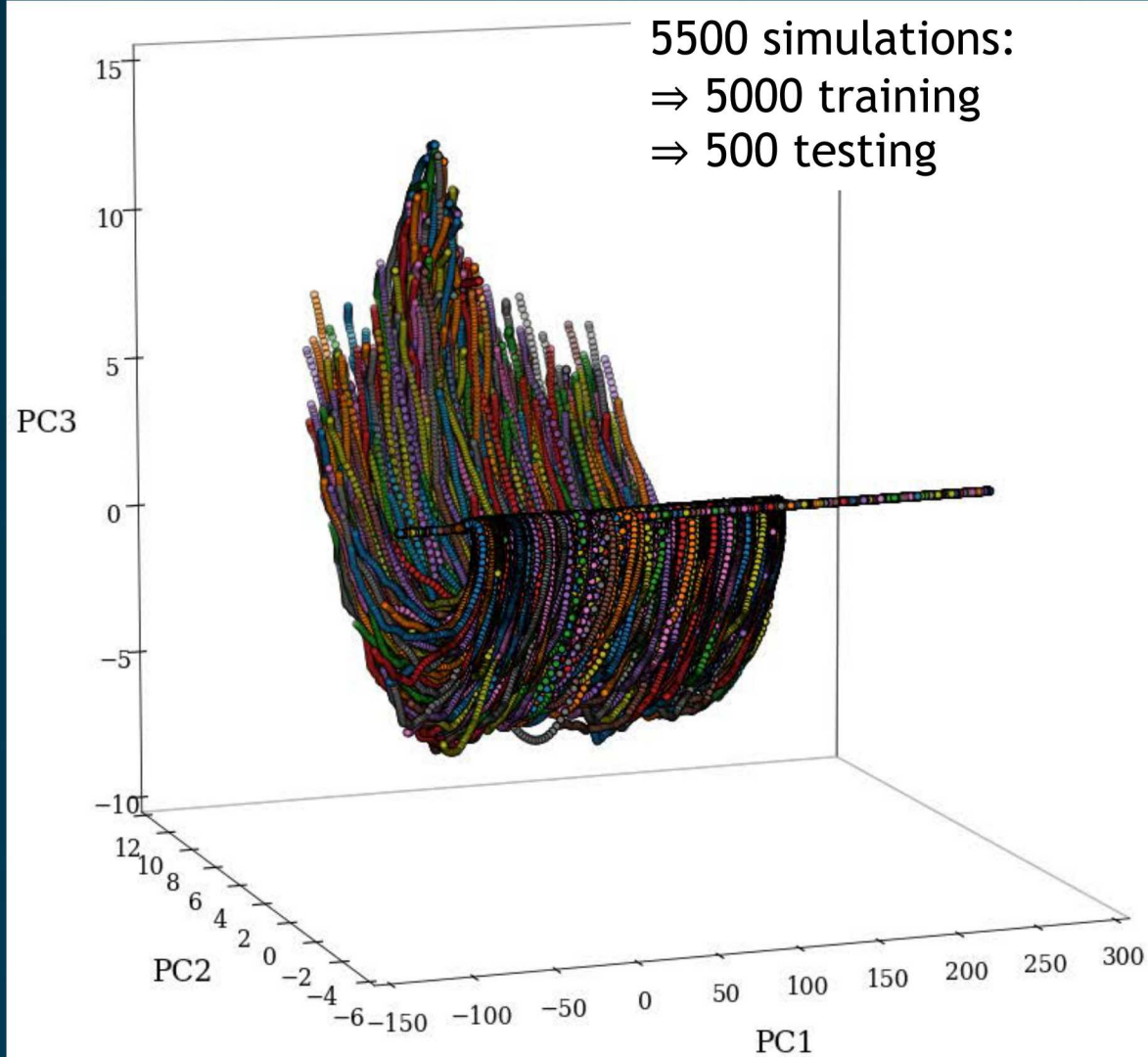
2. Reduction of microstructural space



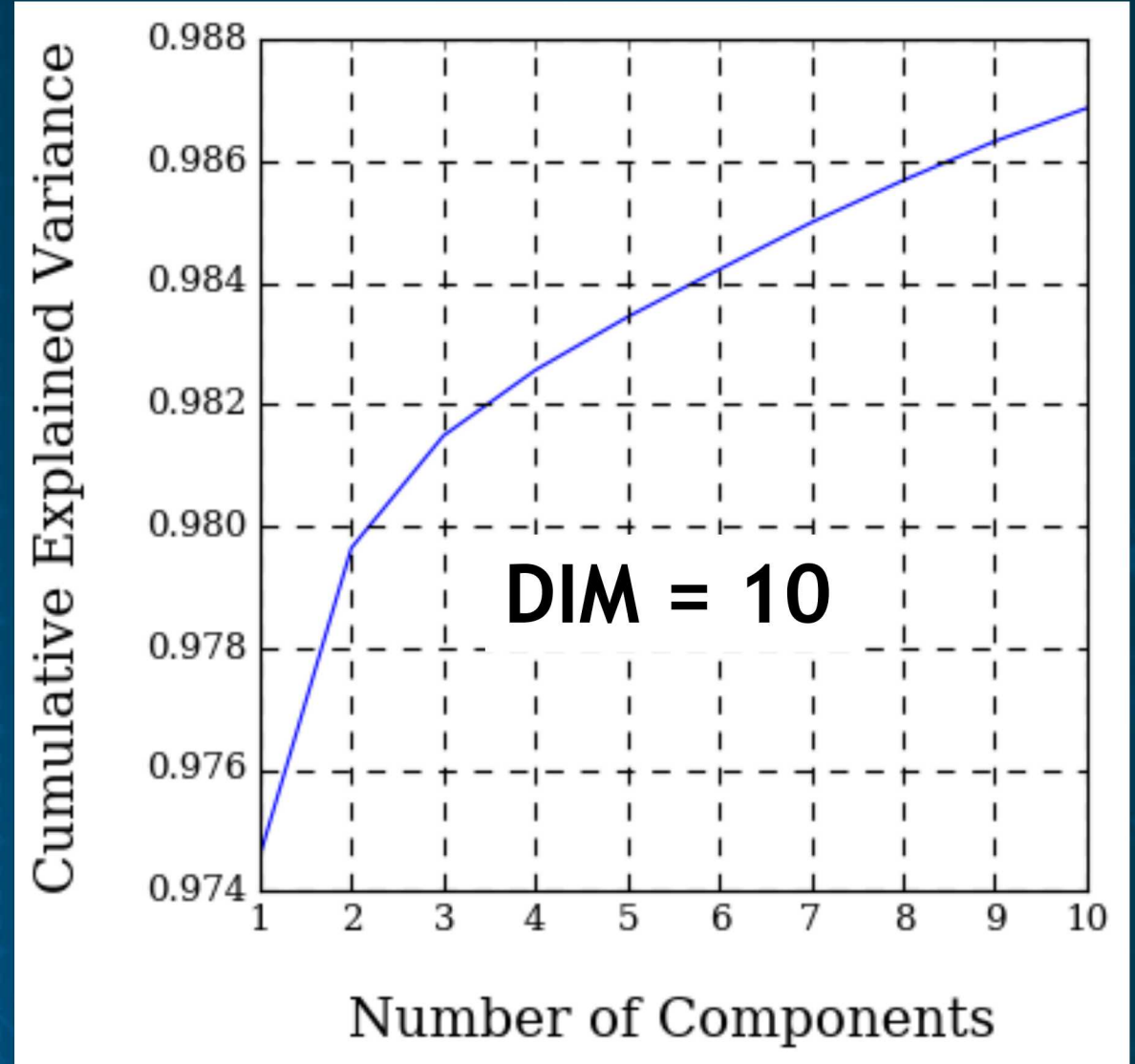
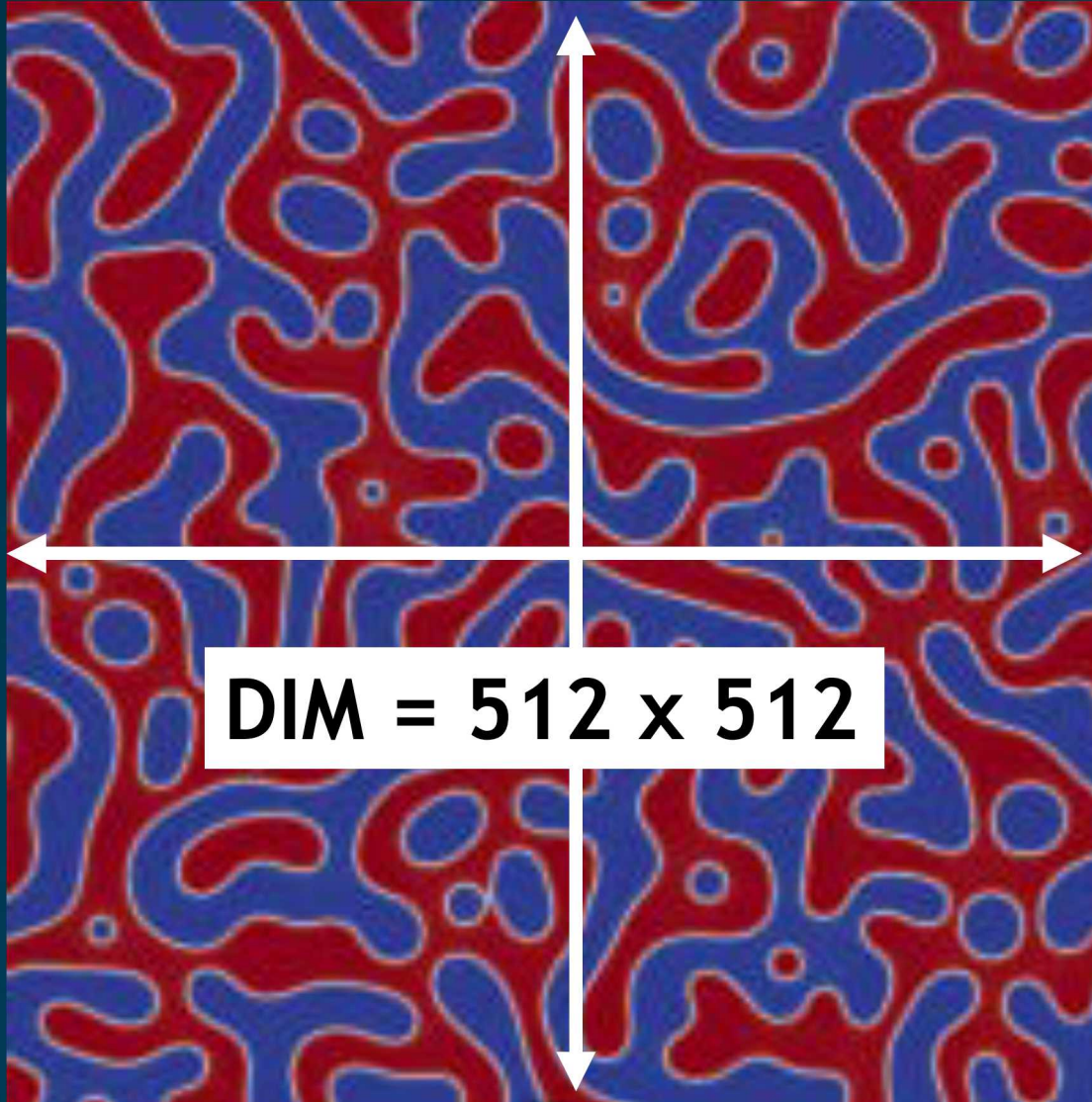
MICROSTRUCTURE TRAJECTORY IN PC SPACE



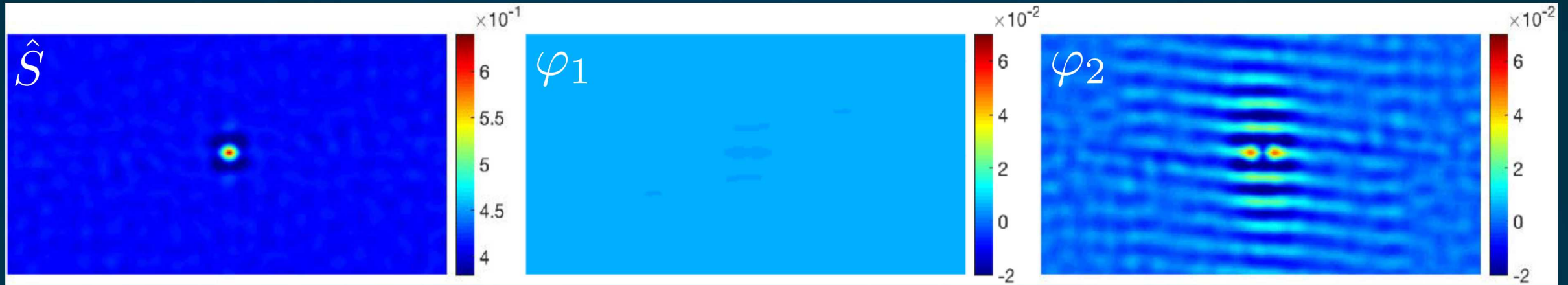
MICROSTRUCTURE TRAJECTORY IN PC SPACE



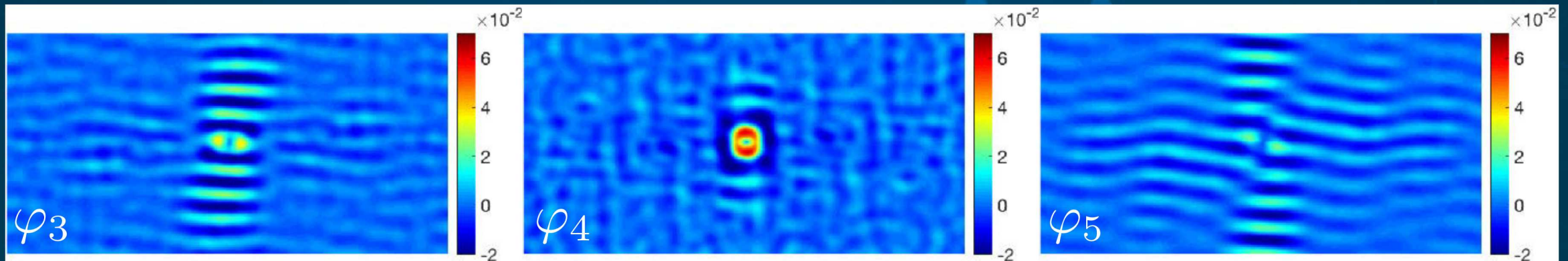
DIMENSIONALITY REDUCTION



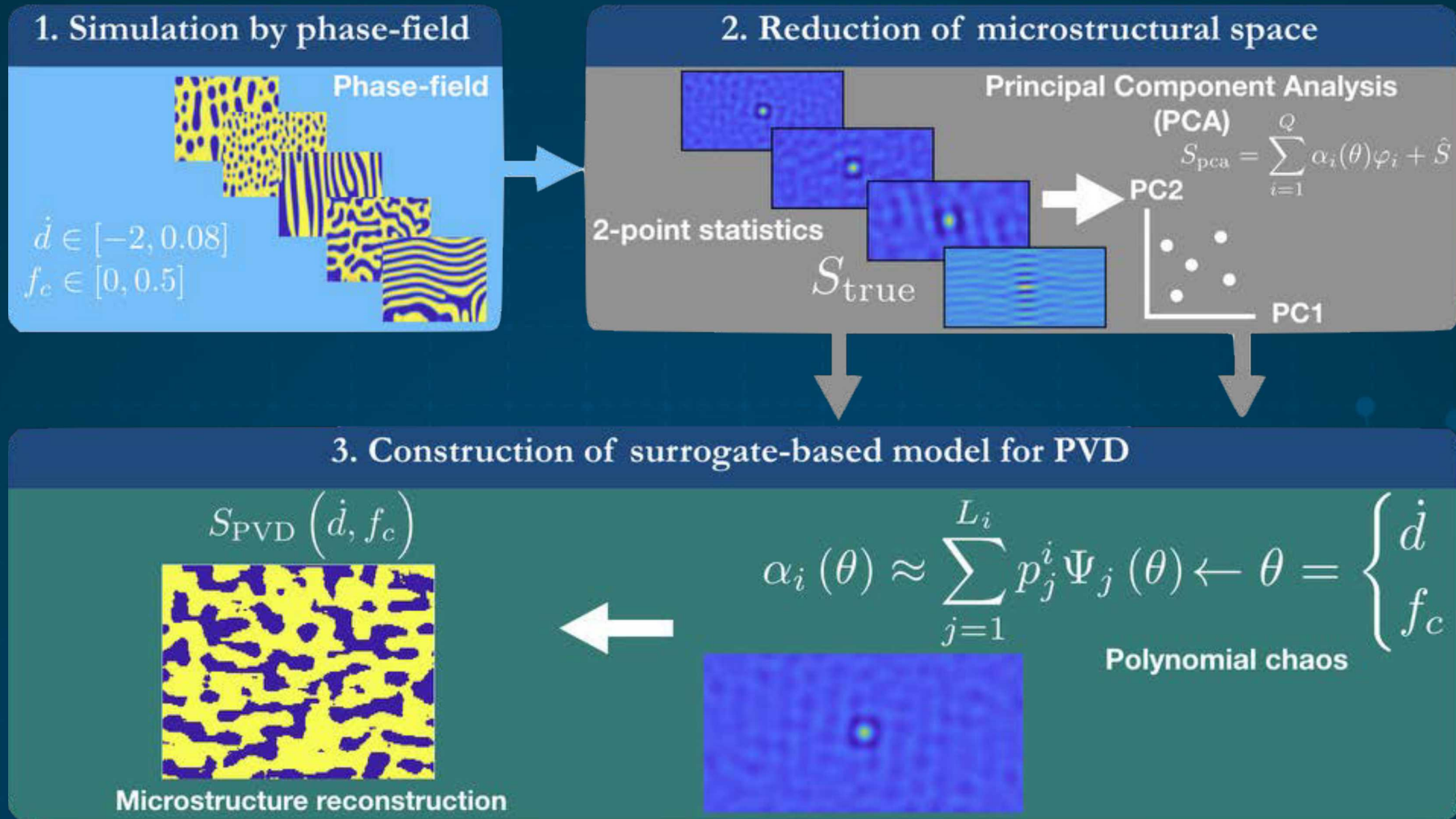
FINGERPRINT OF THE MICROSTRUCTURE?



$$S_{pca}(\theta) = \sum_{I=1}^Q \alpha_i(\theta) \varphi_i + \hat{S}$$

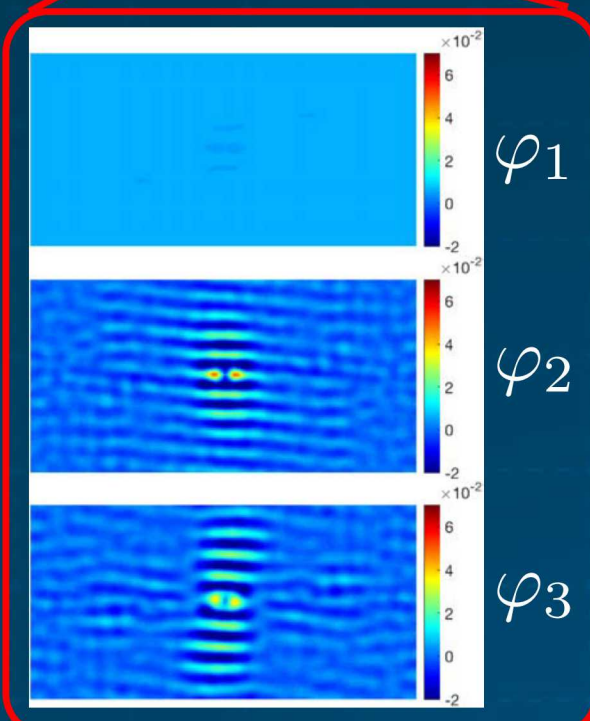


RAPIDLY PREDICTING MORPHOLOGY?



BEYOND CLASSICAL REGRESSION?

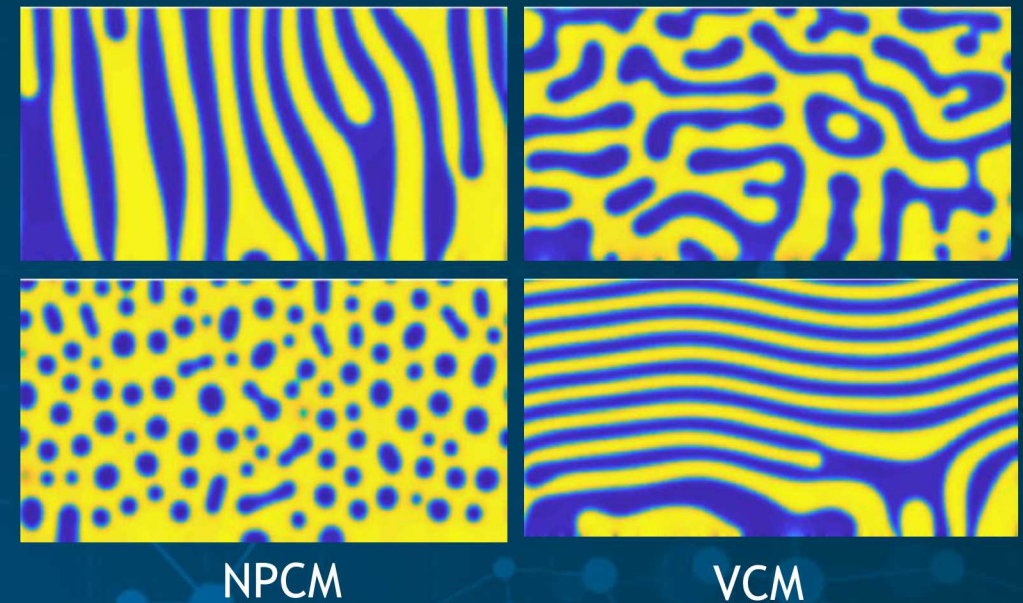
PCA reduces the dimensionality

$$S_{\text{pca}} = \sum_{i=1}^Q \alpha_i(\theta) \varphi_i + \hat{S}$$


Basis elements capture complexity of morphology and long-range correlations

PCE maps back to processing parameters

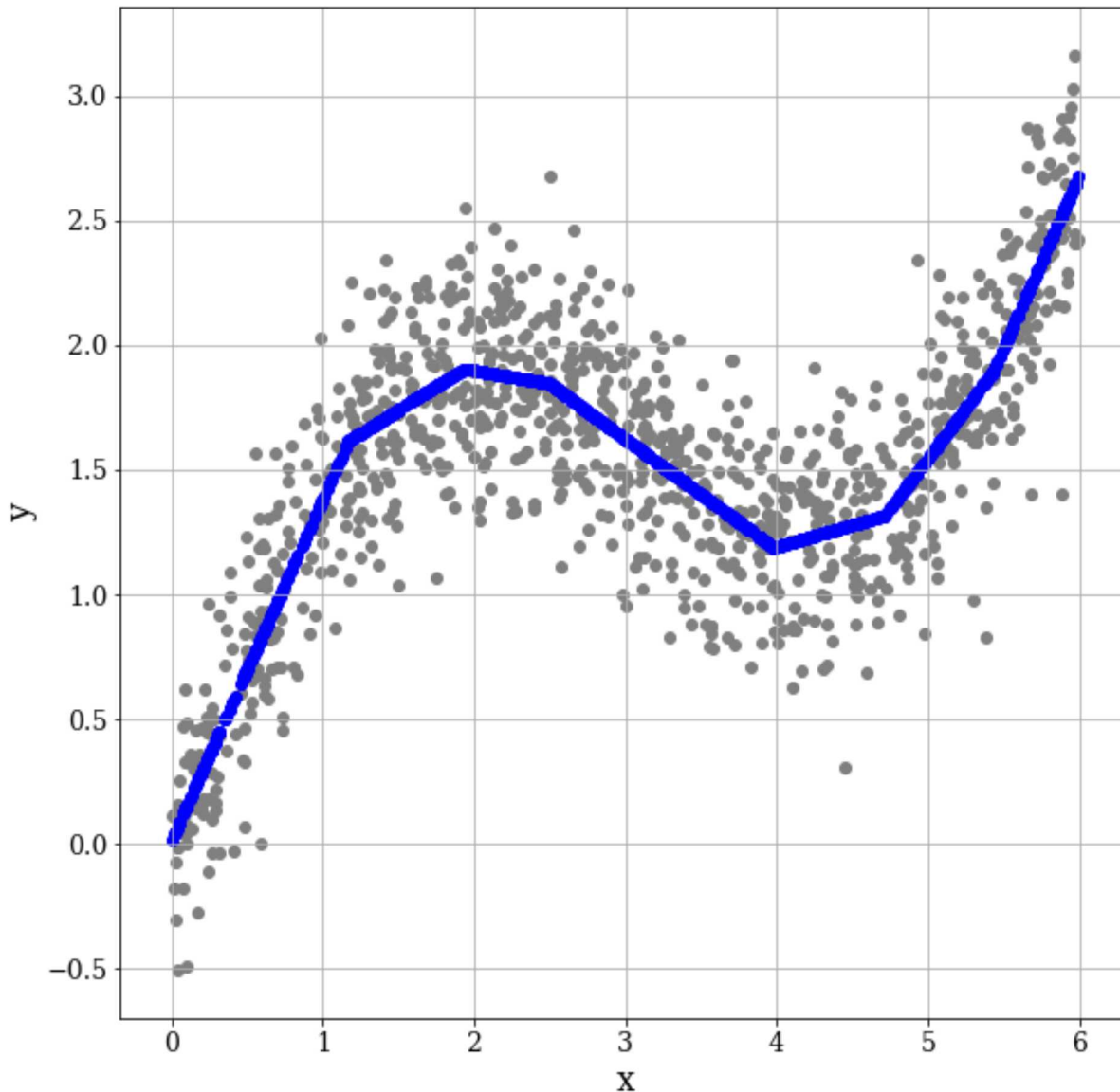
$$S_{\text{predicted}} = \sum_{i=1}^Q \sum_{j=1}^{Li} p_j^i \Psi(\theta) \varphi_i + \hat{S}$$



Small changes in processing input space results in drastic changes in morphology: but...Legendre polynomial in PCE are continuous polynomials

CAN WE LEAP IN TIME?





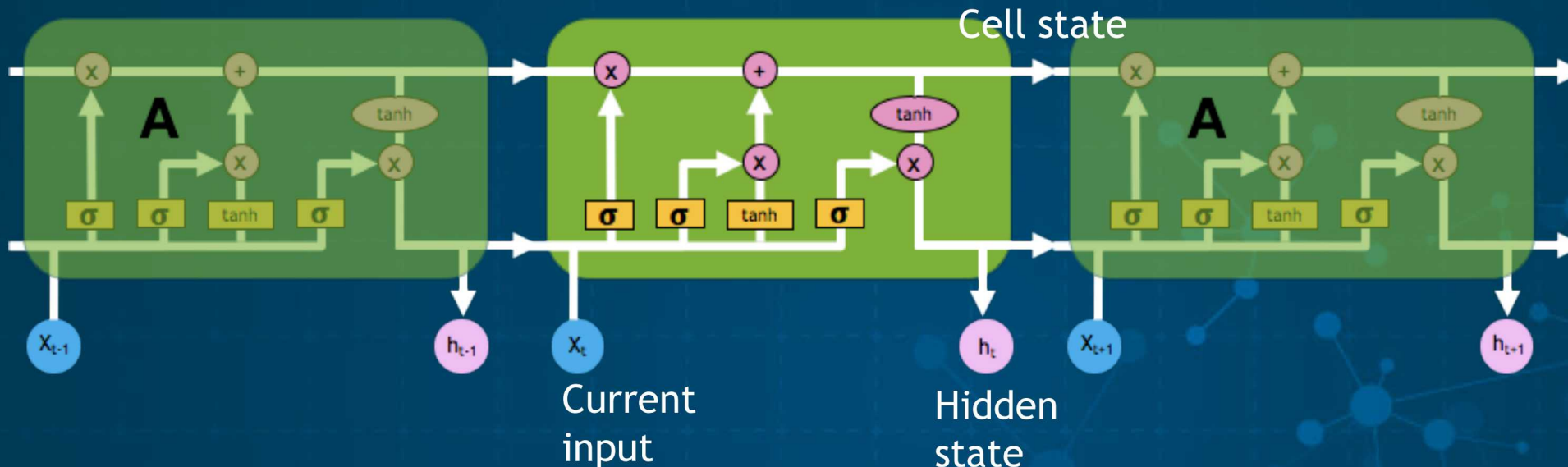
TIME-SERIES MULTIVARIATE ADAPTIVE REGRESSION SPLINES (TSMARS)

1. Useful for identifying nonlinear structure in time series
2. Non-parametric
3. Autoregressive: divide time-series into optimal subdomains to fit splines
4. Predicting microstructure evolution trajectories in PC space
5. Use m most recent time steps to predict $N+1$, $N+2, \dots$

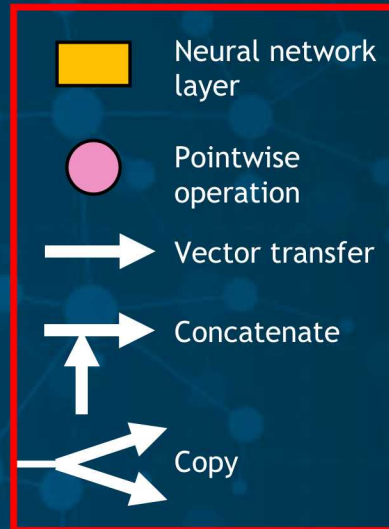
TIME-SERIES FORECASTING

LONG SHORT TERM MEMORY NETWORK (LSTM)

1. LSTM is a special kind of a recurrent neural net (RNN):
Network with loops in them allowing information to persist (i.e., memory)
2. Looping connect previous information to current: TIME HISTORY!
3. Uses previous states, and current input
4. Learn/forget gate (internal structure) used to form long-term memory (known as cell state) and short-term memory (hidden state)
5. Predicting microstructure evolution trajectories in PC space
6. Use entire history to predict $N+1, N+2, \dots$

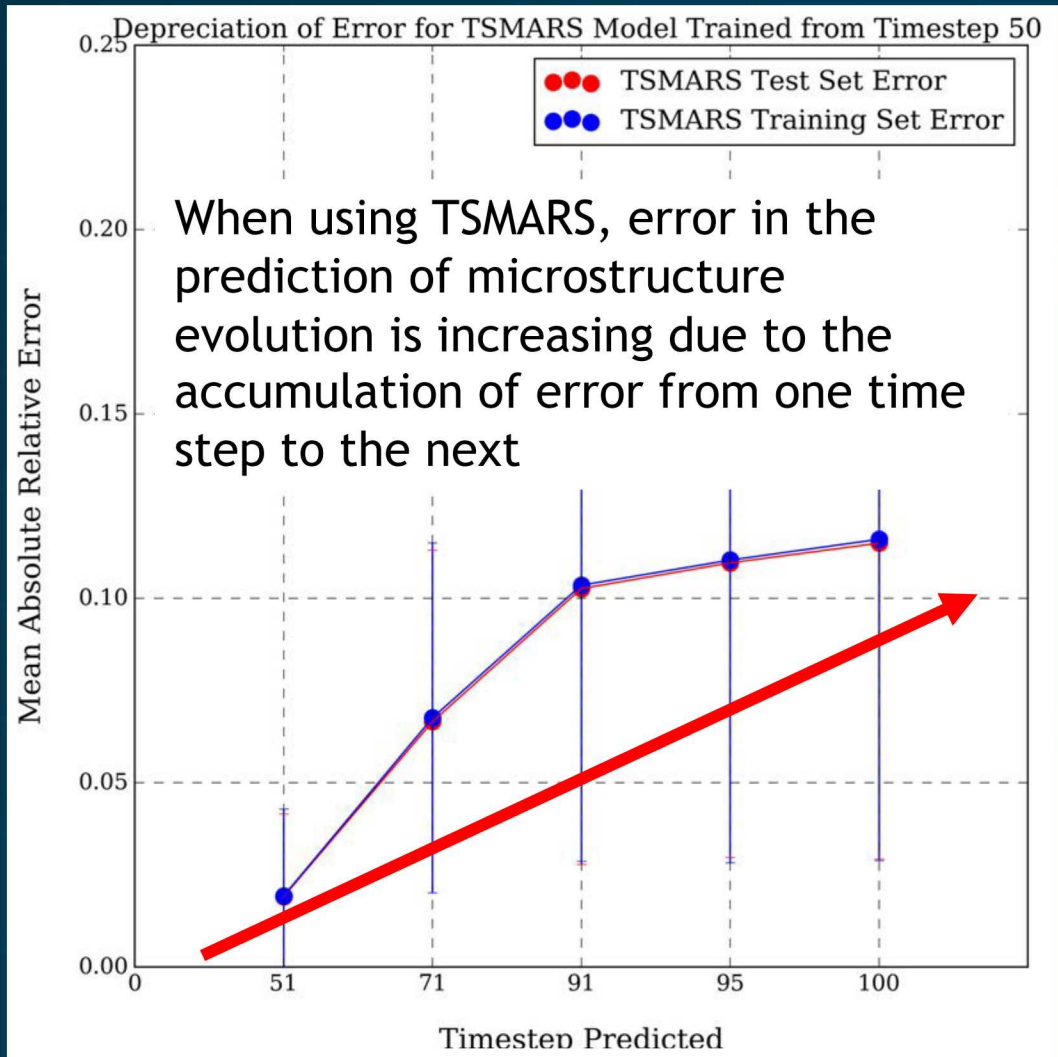


Nomenclature



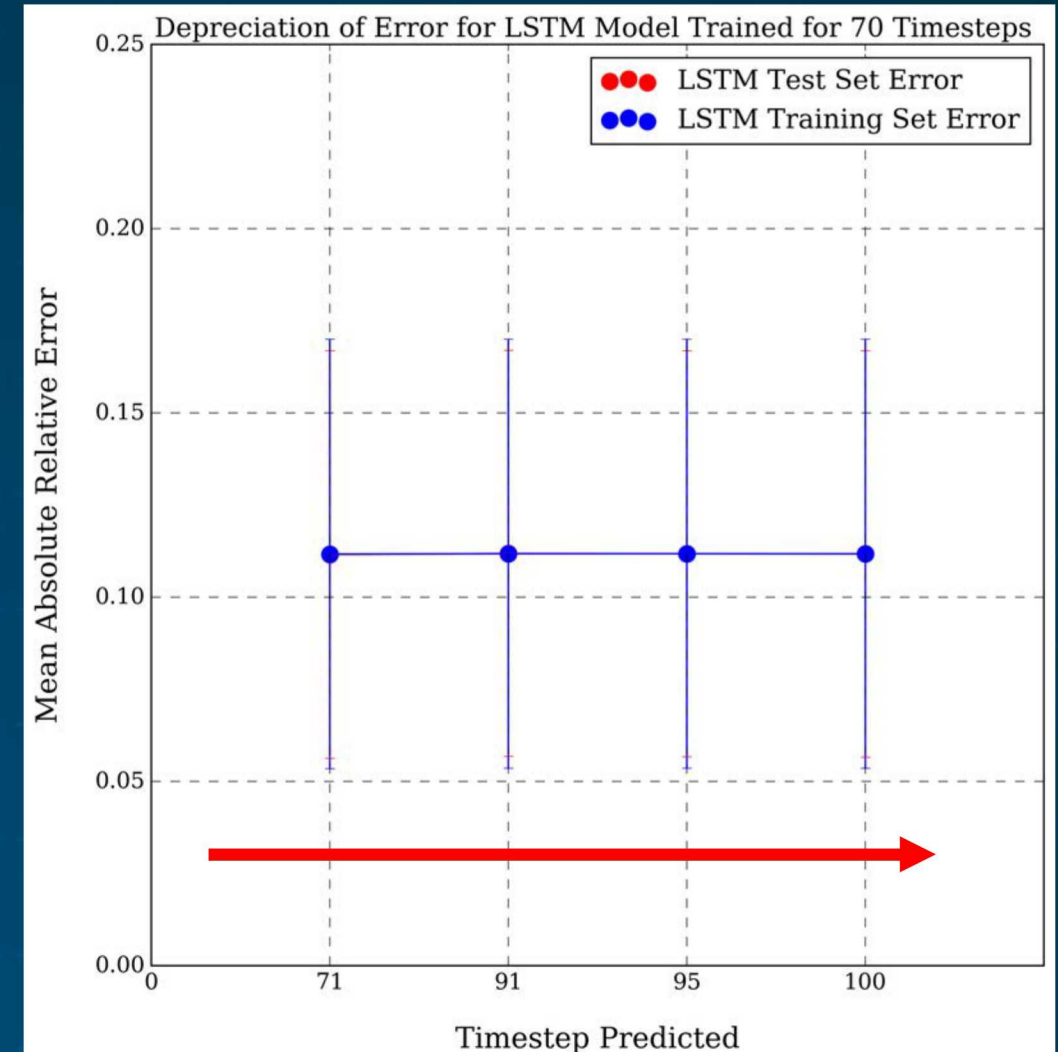
WHY DO WE NEED A DEEP LEARNING STRATEGY?

TSMARS



TSMARS for data augmentation?

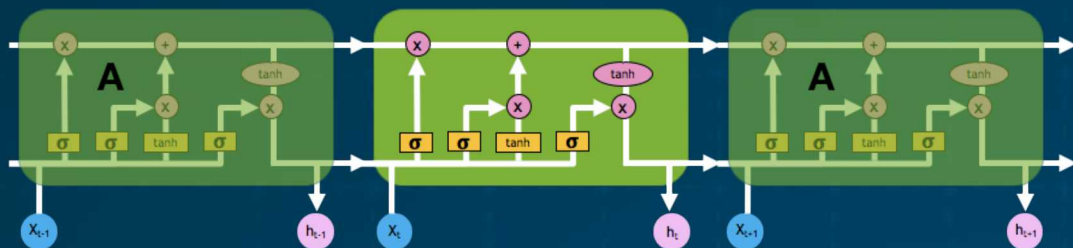
LSTM



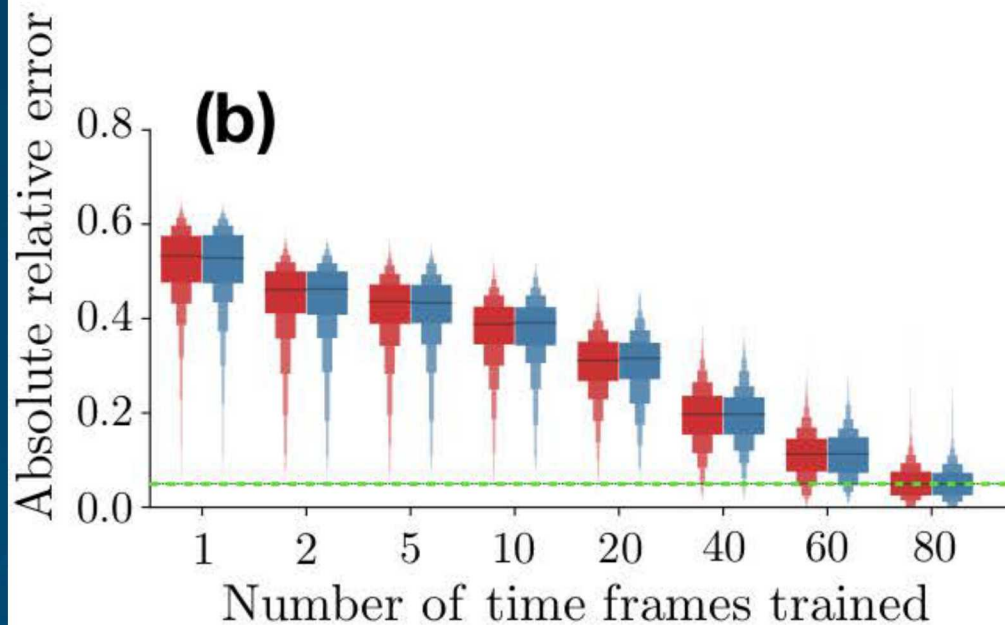
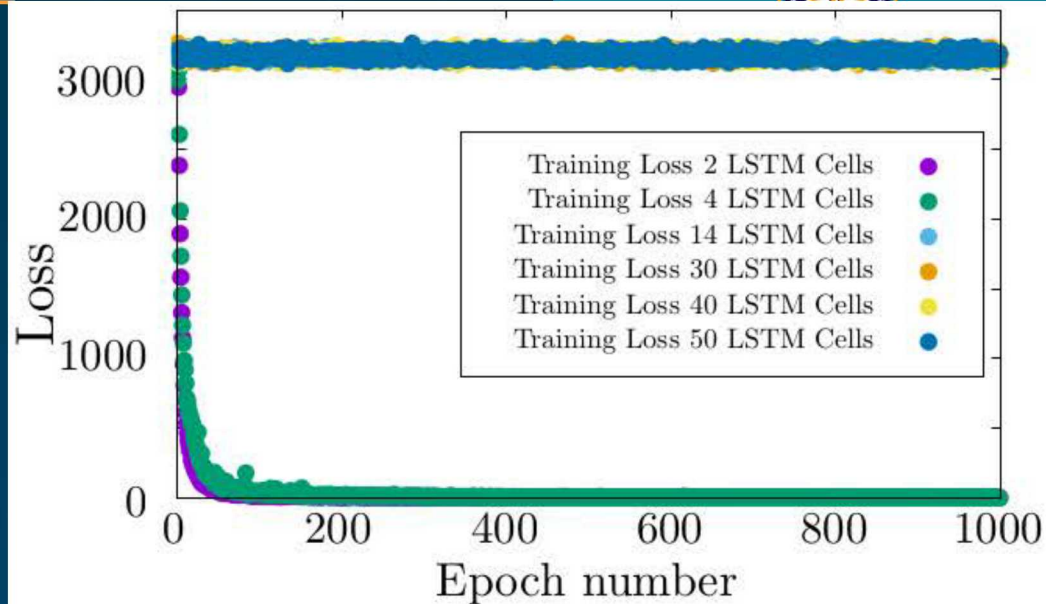
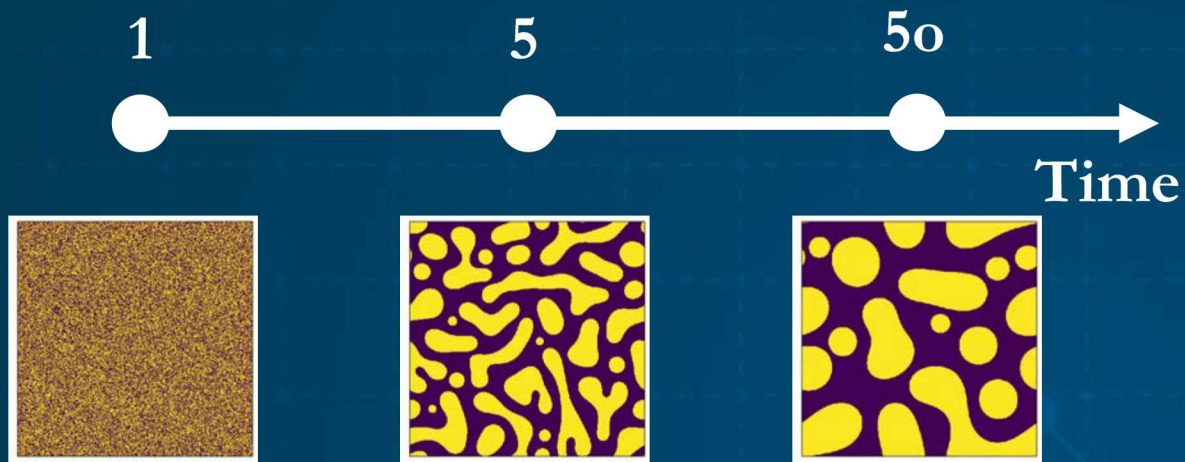
LSTM for PF acceleration?

OPTIMUM LSTM ARCHITECTURE

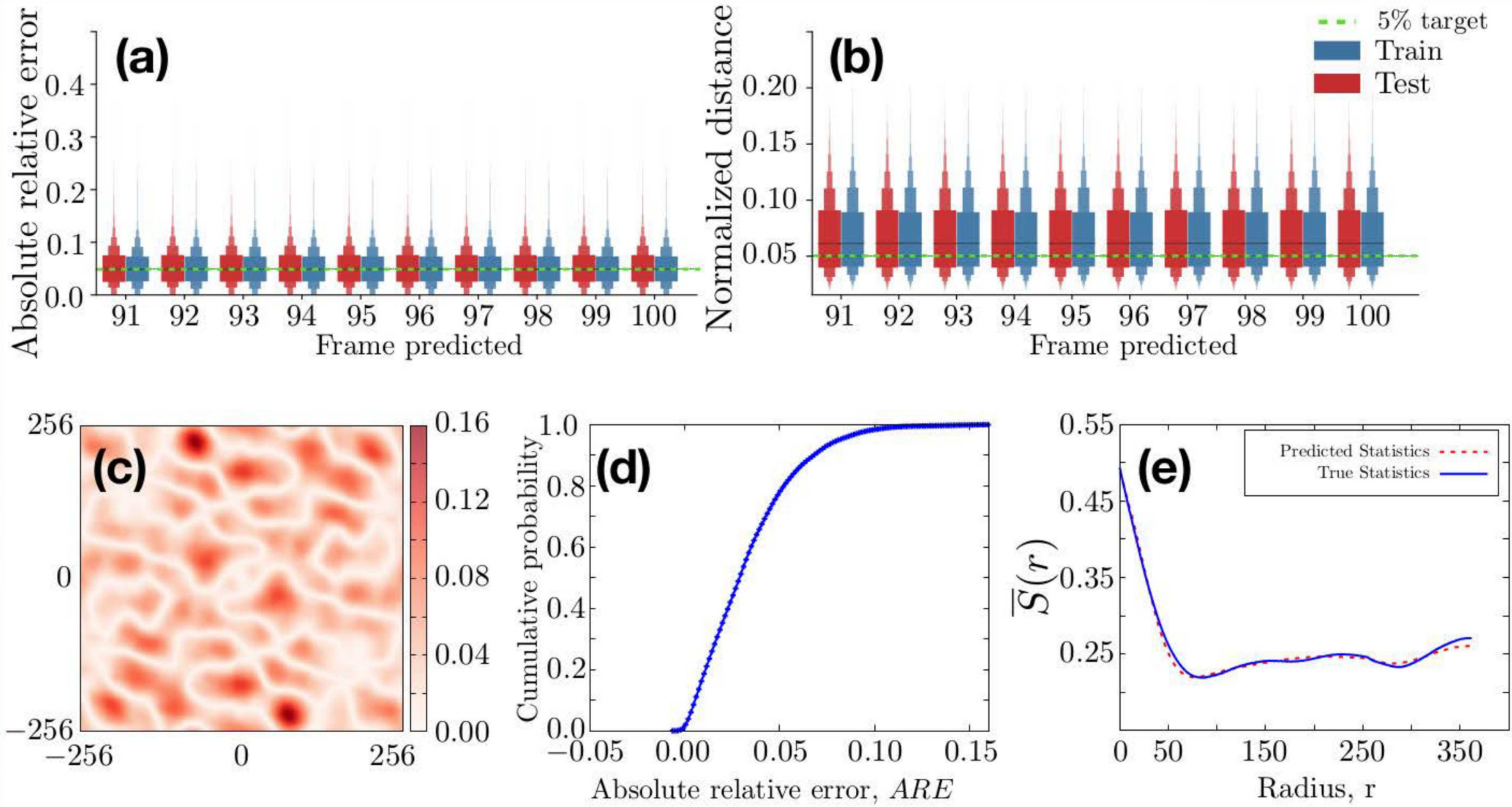
Number of LSTM cells:



Number of time frames needed for training:



PERFORMANCE OF LSTM-TRAINED MODEL



COMPUTATIONAL EFFICIENCY

	High-fidelity	LSTM-trained
1 simulation for 5M time steps	96 mins	0.06s
50,000 simulations	~9 year and 1 month	50 mins



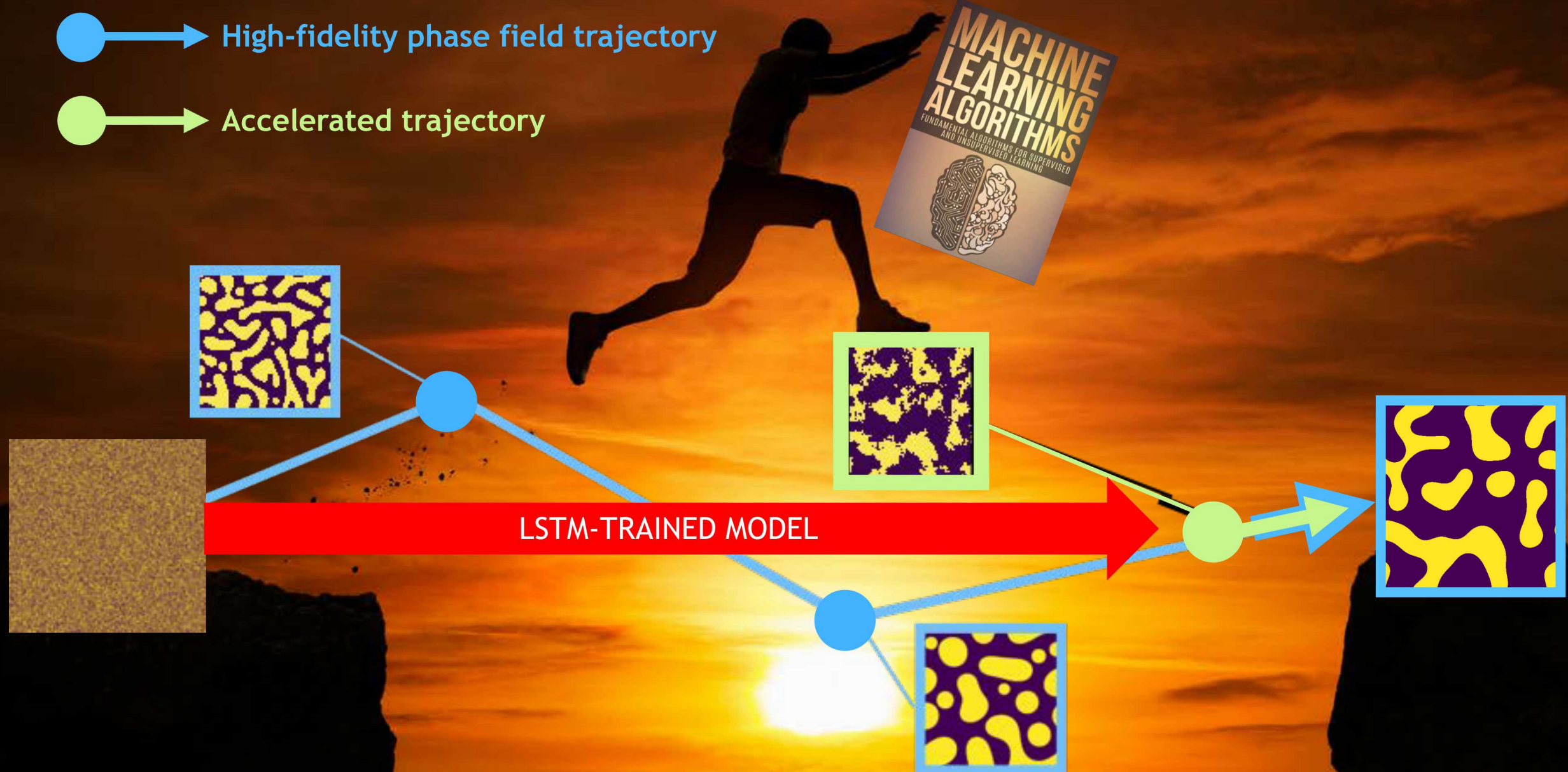
Our LSTM-trained framework is nearly 50,000 faster than the high-fidelity simulation with comparable accuracy!

Opens a promising path forward for discovering, understanding and predicting evolutionary microstructural phenomena

ACCELERATING MICROSTRUCTURE EVOLUTION PREDICTIONS: LEAPING IN TIME USING MACHINE-LEARNED MODEL

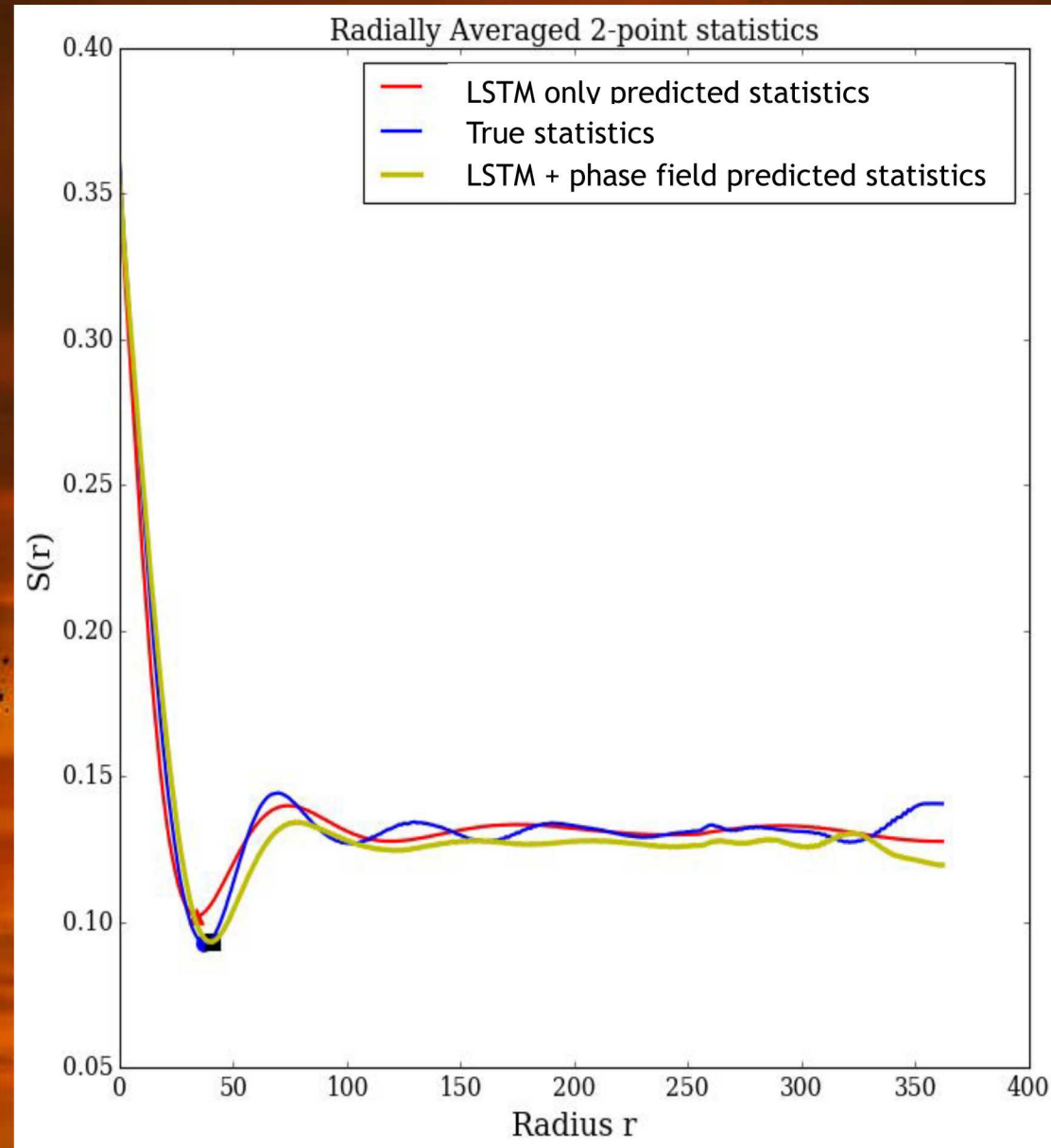
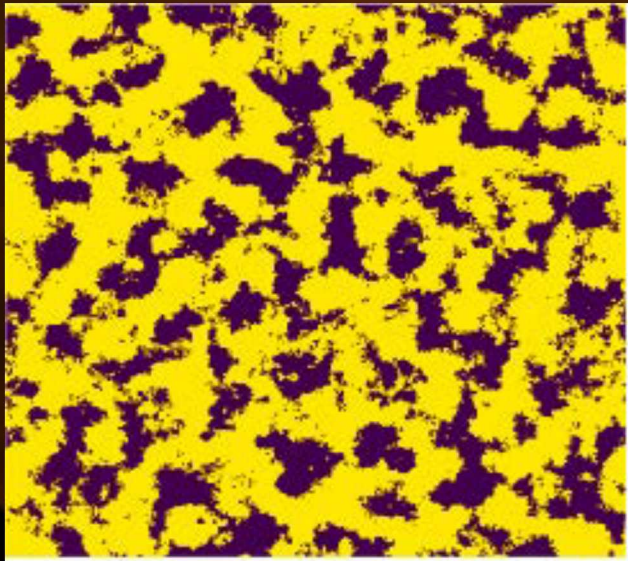
● → High-fidelity phase field trajectory

● → Accelerated trajectory

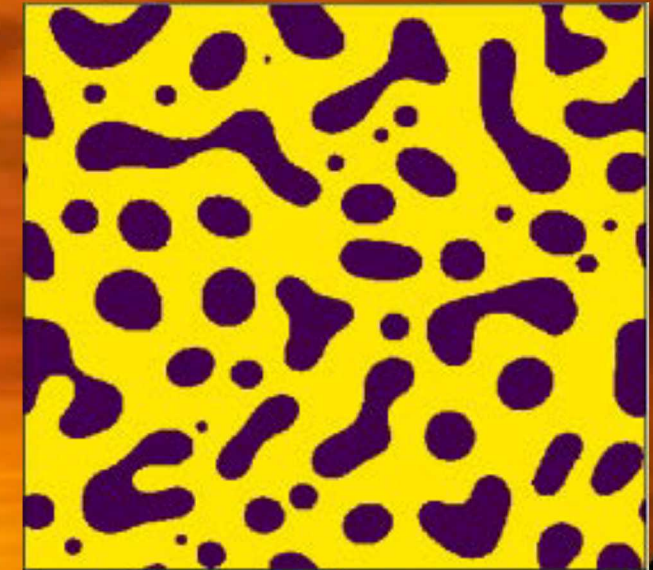


ACCELERATING MICROSTRUCTURE EVOLUTION PREDICTIONS: LEAPING IN TIME USING MACHINE-LEARNED MODEL

LSTM only

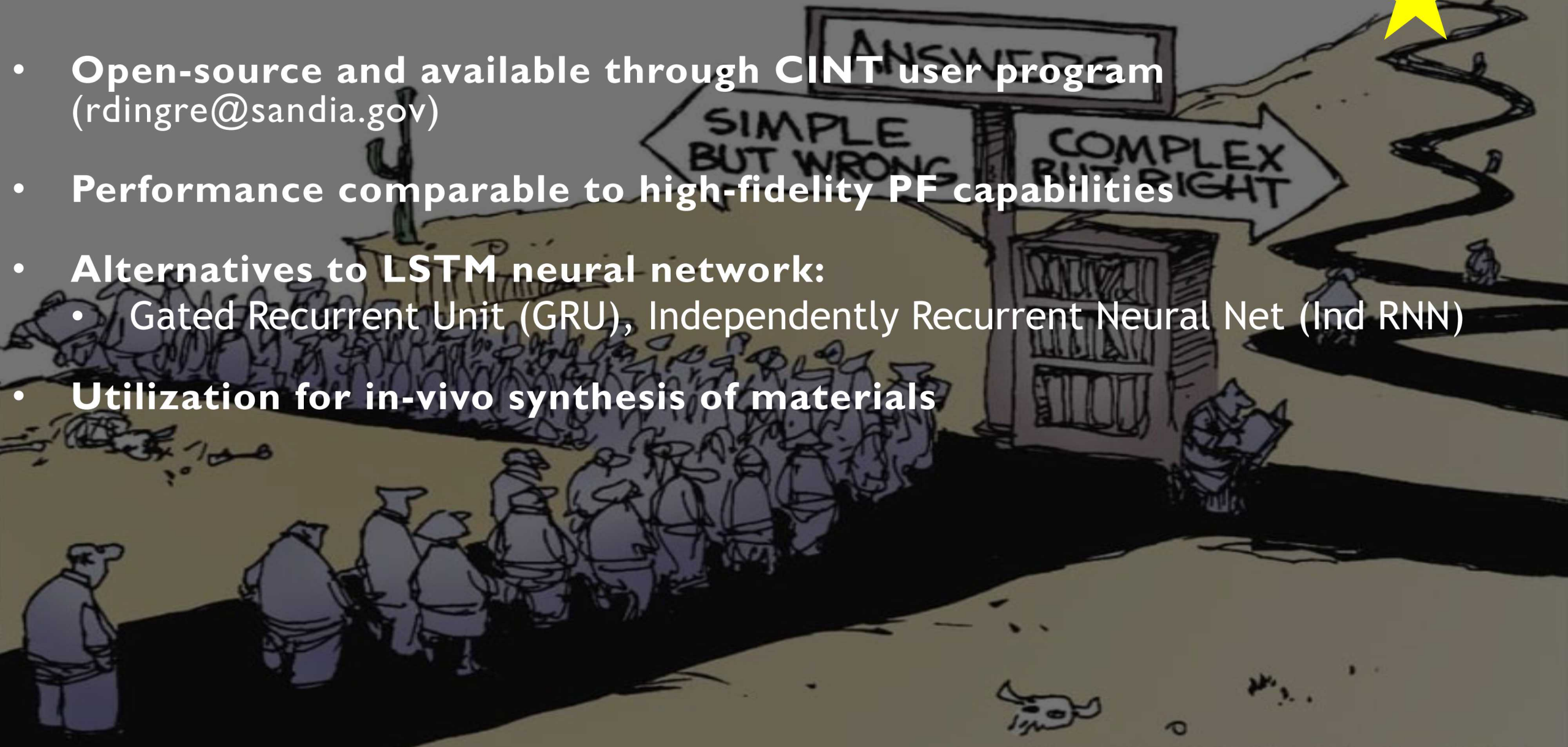


LSTM + phase-field



MOVING FORWARD

- **Open-source and available through CINT user program**
(rdingre@sandia.gov)
- **Performance comparable to high-fidelity PF capabilities**
- **Alternatives to LSTM neural network:**
 - Gated Recurrent Unit (GRU), Independently Recurrent Neural Net (Ind RNN)
- **Utilization for in-vivo synthesis of materials**





FREE SCIENCE...

rdingre@sandia.gov

CINT is a user facility providing cutting-edge nanoscience and nanotechnology capabilities to the research community.

Access to our facilities and scientific expertise is **FREE** for non-proprietary research.

Research areas:

- Quantum Materials Systems
- Nanophotonics and Optical Nanomaterials
- In-Situ Characterization and Nanomechanics
- Soft, Biological, and Composite Nanomaterials

To learn more and
apply to use the facilities, visit:
<https://cint.lanl.gov>

