

Multi-Junction Solar Cells Paving the Way for Super High-Efficiency

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In order to realize a clean energy society by using renewable energies, high-performance solar cells are very attractive. The development of high-performance solar cells offers a promising pathway toward achieving high power per unit cost for many applications. As state-of-the-art of single-junction solar cells are approaching the Shockley-Queisser limit of 32-33%, an important strategy to raise the efficiency of solar cells further is stacking solar cell materials with different bandgaps to absorb different colors of the solar spectrum. The III-V semiconductor materials provide a relatively convenient system for fabricating multi-junction solar cells providing semiconductor materials that effectively span the solar spectrum as demonstrated by world record efficiencies (39.2% under 1-sun and 47.1% under concentration) for 6-junction solar cells. This success has inspired attempts to achieve the same with other materials like the perovskites for which lower manufacturing costs may be achieved. Recently, Si MJ solar cells such as III-V/Si, II-VI/Si, chalcopyrite/Si and perovskite/Si have become popular and are getting closer to economic competitiveness.

Here, we discuss perspectives of MJ solar cells from the viewpoint of efficiency and low-cost potential based on scientific and technological arguments and possible market applications. In addition, this article provides a brief overview of recent developments with respect to III-V MJ solar cells, III-V/Si, II-VI/Si, Perovskite/Si tandem solar cells and some new ideas including so-called 3rd generation concepts.

I. INTRODUCTION

The development of high-performance solar cells offers a promising pathway toward achieving high power per unit cost for many applications. Various single-junction solar cells have been developed and efficiencies of 29.1%, 26.7%, 23.4%, 22.1% and 21.6% (small area efficiency of 25.2%) have been demonstrated¹ with GaAs, Si, CIGSe, CdTe and perovskite solar cells, respectively. However, single-junction solar cells may be capable of attaining AM1.5 efficiencies of up to 30-32% as shown in Fig. 1². That is, state-of-the-art of single-junction solar cells are approaching the Shockley-Queisser limit³. An important strategy to raise the efficiency of solar cells is stacking solar cell materials with different bandgaps to absorb different colors of the solar spectrum. This so-called ‘multi-junction (MJ)’^{4,5} approach can reduce the thermalization loss due a high-energy photon absorbed by a small-bandgap material, and below-bandgap loss due to a low-energy photon of insufficient energy to excite an electron in a high-bandgap material as shown in Fig. 2⁶. Figure 3 shows the principle of wide photoresponse using MJ solar cells for the case of a triple-junction solar cell⁷. Solar cells with different bandgaps are stacked one on top of the other so that the solar cell facing the sun has the largest bandgap (in this example, this is the GaInP top solar cell with bandgap energy E_g of 1.8-1.9 eV). This top solar cell absorbs all the photons at and above its bandgap energy and transmits the less energetic photons to the solar cells below. The next solar cell in the stack (here the GaAs middle solar cell with E_g of 1.42 eV) absorbs all the transmitted photons with energies equal to or greater than its bandgap energy, and transmits the rest downward in the stack (in this example, to the Ge bottom solar cell with E_g of 0.67 eV). Of all the so-called third generation solar cell strategies⁸, only MJ designs have been successful to surpass the detailed-balance limit of single-junction solar cells. Such successful achievements are thought to be due to longtime R&D since late 70s, bandgap engineering including lattice matching, high quality epitaxial growth and so forth.

The operating principles of MJ solar cells were suggested by Jackson⁹ as long ago as 1955, and they have been investigated since 1960¹⁰. This concept was most successfully implemented in III–V compound semiconductor solar cells, since a compound semiconductor has a good range of lattice parameters and bandgaps to choose from. High efficiencies of 32.8%¹ under 1-sun and 35.5%¹¹ under concentration with 2-junction solar cells, 37.9%¹² under 1-sun and 44.4%¹² under concentration with 3-junction solar cells, 46.1%¹³ under concentration with 4-junction solar cell, 38.8%¹⁴ under 1-sun with 5-junction solar cell, 39.2%¹⁵ under 1-sun and 47.1%¹⁵ under concentration with 6-junction solar cells have been demonstrated as shown in Fig. 4. Figure 4 shows the chronological improvement in conversion efficiency^{7,16} of concentrator MJ and one-sun MJ solar cells

in comparison with those of crystalline Si, GaAs, CIGS and perovskite single-junction solar cells.

Recently, Si based tandem solar cells¹⁶ such as III-V/Si^{17,18}, II-VI/Si¹⁹, chalcopyrite/Si²⁰, CZTS/Si²¹ and perovskite/Si²² tandem solar cells are expected to play a more important role as high-efficiency, low-cost solar cells move closer to industrial manufacturing. In addition, there are other approaches such as perovskite/perovskite²³, III-V/CIGSe²⁴, perovskite/CIGSe²⁵ MJ solar cells which are still at a lower technology readiness level but may become very attractive candidates for photovoltaic energy conversion in the future.

Here, we discuss perspectives of MJ solar cells from the viewpoint of efficiency and low-cost potential based on scientific and technological arguments and possible market applications. In addition, this article provides a brief overview of recent developments with respect to III-V MJ solar cells, III-V/Si, II-VI/Si, Perovskite/Si tandem solar cells and some new ideas including so-called 3rd generation concepts⁸.

II. SCIENTIFIC CONSIDERATION

The fundamental processes in photovoltaic power conversion are shown in Figure 5, incident sunlight of energy above the semiconductor bandgap can be absorbed (1) and excess energy dissipated as a thermalization loss (3); photons below the bandgap energy can pass through the solar cell unabsorbed (2). Excess radiative recombination proceeds due to the presence of photogenerated carriers (4). At forward operating voltages, the free energy of the carriers is determined by the quasi-Fermi level separation which defines the solar cell voltage at the electrical contacts to the solar cell, $V = \mu_e - \mu_h$.

The breakdown between power generated by the solar cell and these losses is illustrated in Figure 2⁶. For a single-junction solar cell, the two largest losses are the thermalization and below-E_g losses, both of which are significantly mitigated with the addition of semiconductor junctions with different bandgap energies in a MJ device. This is because a larger portion of the solar spectrum is then absorbed close to the bandgap of one of the semiconductors and therefore experiences less thermalization of carriers. All other fundamental losses *increase with increased number of semiconductor junctions and are discussed in detail elsewhere³. We note that all three remaining losses

* Note that resistance losses are not regarded as fundamental losses, though they are unavoidable in practical devices and in fact benefit from lower currents which are typically achieved with more junctions.

fundamentally depend on solar cell temperature and can therefore be reduced by operating the solar cell at a lower temperature:

The so-called ‘Boltzmann loss’ is an entropic loss associated with the increase in the occupancy of optical modes on re-emission of light that results in a voltage loss. Practically it can be recovered in two ways, conventionally by increasing the solar concentration on the solar cell, or equivalently, restricting the radiative emission from the cell. Dividing up the solar spectrum between an increasing number of junctions results in emission at multiple wavelengths and hence a larger Boltzmann loss. The loss can be mitigated by paying attention to the geometrical optical arrangement, for example through solar concentration or angular restriction of radiative recombination.

The Carnot loss arises from establishing carriers at finite temperature in a band and hence rises as further junctions establish additional bands occupied by photogenerated carriers. The loss cannot be recovered except in very unusual circumstances where the PV cell is operated at low temperature.

The Emission loss is unavoidable in a conventional solar cell owing to the reciprocity²⁶ between absorption and emission encapsulated by Kirchhoff’s law of radiation. The detailed balance limit³ considers only this unavoidable radiative recombination loss through thermodynamic arguments. The external radiative efficiency (ERE) describes how closely a subcell comes to this thermodynamic limit as all other non-radiative recombination is considered potentially avoidable. In an optimally configured MJ solar cell, the emissive loss is small, but in a series connected tandem solar cell where current flow is constrained by one sub-cell, there can be significant transfer of energy down the tandem absorber stack. This is known as radiative coupling and discussed in more detail below. An extreme case of radiative coupling arises if the reciprocity between absorption and emission is lifted, potentially using magneto-optical materials²⁷, that allows the efficiency of an infinite tandem stack (asymptotes to 86.8%) to be raised to the Landsberg efficiency of 93.5%. At the Landsberg limit, each of the component junctions operates arbitrarily close to V_{oc} and electrical power delivered from the solar cell infinitesimally slowly²⁸.

To achieve efficient operation, the photogeneration rates in a series connected solar cell should be closely matched. While the choice of bandgap and hence absorption threshold for the component junctions plays a primary role, the sub-cell photogeneration can be optimized by adjusting the thickness of the junctions such that an overperforming sub-cell can allow some light to pass unabsorbed into an underlying junction²⁹. This approach works well for static solar spectra (such as AM0) but for the terrestrial spectrum, the spectral irradiance varies throughout the day and between seasons with noticeable

effects on system performance³⁰. If the semiconductor material is radiatively efficient, sub-cells with excessive photogeneration will radiate the excess into lower lying junctions, with a small fraction of this escaping from the top of the solar cell. Radiative dominated behaviour has been observed in III-V solar cells³¹ and some perovskite materials³², even appearing as a measurement artefact in MJ devices^{33,34}. Since radiative coupling transfers energy down the MJ stack, the effect can mitigate the effects of spectral mismatch under blue-rich spectral conditions³⁵ as well as offering some freedom in tolerable absorber bandgap configuration³⁶ in particular enabling efficient operation of higher-gap top solar cells in a silicon based tandem³⁷.

III. BRIEF OVERVIEW

3.1. III-V MJ

III-V semiconductor materials have many advantages for high-efficiency solar cells in general and MJ solar cell in particular. III-V semiconductors consist of elements from the group III (Al,Ga,In) and V (N,P,As,Sb) columns in the periodic table arranged in a zinc-blend (or wurtzite) crystal structure. The highest single-junction efficiency has consistently remained a GaAs solar cell due to its bandgap match to the solar spectrum and its high ERE as shown in Fig. 1. ~~ERE is expressed by eq. (1) and the solar cells with the higher ERE value show the less of non-radiative recombination loss.~~ III-V alloys (hereafter III-Vs) cover a wide bandgap range from 2.4 eV down to almost 0.0 eV as shown in Fig. 6, with III-N alloys covering much higher bandgaps³⁸ (not shown).

Many of the III-Vs can be grown as *single-crystal layers* on single-crystal substrates (e.g. GaAs, InP, Ge etc.) using liquid phase epitaxy (LPE), molecular beam epitaxy (MBE), organometallic vapor phase epitaxy (OMVPE), and hydride vapor phase epitaxy (HVPE) techniques. Many III-Vs *can be doped* both n-type and p-type over a wide density range and are direct-gap for nearly complete thin-film absorption. They can have *high mobilities*, relatively *long lifetimes*, and *low interface recombination* when higher bandgap alloys are used for heterojunction passivation (i.e. window and back-surface-field layers) that result in *diffusion lengths* longer³⁹ than the required single-pass absorption thicknesses. Thus, light trapping techniques often used in silicon photovoltaics have not been much required for III-V materials. The primary disadvantage of III-V solar cells is their *sensitivity to defects* that act as deep non-radiative recombination sites, such as dislocations, impurities, and phase boundaries, but they also have a relatively low density of intrinsic native defects. While III-Vs can be grown with a high degree of crystal perfection that avoids the problems of defects, it can only presently be done at a relatively

high *cost* that results in the other primary disadvantage of III-V solar cells. Due to the high cost of III-V solar cells fabrication, they have been used in applications that leverage the efficiency advantage on the balance of system costs, such as space, concentrator, and other area or weight constrained applications. Concentrator applications have also been intimately tied to III-V MJ development because of the logarithmic increase in voltage with concentration for each junction.

From the beginning of MJ solar cell research, III-V alloys have been the material of choice due to the advantages listed above. In addition, the wide range of bandgaps appropriate for solar collection at the same lattice constant, as shown in Fig. 6, has provided many avenues for creating defect-free monolithic MJ configurations. Several of the III-V MJ strategies with their achieved efficiencies are summarized in Fig. 6 and mentioned briefly in the following paragraph.

Initially, multi-terminal structures were considered^{40,41}, but since low-resistance tunnel junction interconnects consisting of heavily doped wide-bandgap materials. have been demonstrated in III-V materials^{42,43}, series-connected MJ solar cells have dominated. Such two-terminal devices facilitate the integration of the solar cells into modules and the connection to inverters. Target bandgap combinations for series-connected MJ solar cell have been calculated for specific spectra assuming detailed-balance^{44,45} and more realistic ERE limitations⁴⁶. Recently, wafer bonding¹³, transparent conductive oxide layer insertion⁴⁷ and mechanical stacking¹⁷ for MJ solar cell formation have been demonstrated.

The first III-V MJ strategy considered was the easily lattice-matched AlGaAs/GaAs structure⁴⁸⁻⁵⁰, but difficulties with defects associated with oxygen incorporation into AlGaAs restricted progress⁵¹. The surprising stability⁵² and quality of epitaxially-grown GaInP lattice-matched to GaAs finally resulted in the GaInP/GaAs solar cell with an efficiency of 29.5%⁵³ to exceed the theoretical potential of any single-junction solar cell. Spontaneous ordering in GaInP material also resulted in adjustable bandgaps for the same alloy composition⁵⁴. The use of Ge substrates was primarily introduced to GaInP/GaAs solar cells for the mechanical and cost advantages over GaAs substrates, but the serendipitous introduction of a diffuse Ge junction also resulted in a slight voltage increase without much added cost or complexity. This GaInP/GaAs/Ge three junction (3J) solar cells has remained the standard for space and concentrator applications to this day. Improvements on this 3J solar cell revolved around replacing the low-E_g Ge junction with a higher-E_g III-V junction. Dilute nitride GaInNAs(Sb) that is lattice-matched to GaAs was a very promising 1.0 eV candidate⁵⁵, but sufficient quality was not obtained over many years with industry standard OMVPE growth. Using (arguably) more expensive MBE growth, excellent concentrator 3J solar cells with a dilute nitride bottom junction⁵⁶

have been demonstrated⁵⁷. While high densities of threading dislocations (TDD) are detrimental to III-V quality⁵⁸, metamorphic growth (which allows the mismatched strain to be relaxed slowly in compositionally step graded buffer layers) has allowed high-quality lattice-mismatched InGaAs junctions to be grown with $TDD \sim 1 \times 10^6 \text{ cm}^{-2}$ on GaAs substrates. The inverted metamorphic multi-junction (IMM) strategy⁵⁹ that grows the low bandgap InGaAs junctions last has been demonstrated with three^{60,61}, four^{62,63}, and six junction¹⁵ III-V solar cells with extremely high efficiencies. A natural consequence of the IMM strategy is the removal of the substrate, which has advantages for cost with potential substrate reuse, light-weight, and flexible solar cells^{64,65}, and back surface reflectors for photon recycling^{31,66}. The dislocations in mismatched InGaAs could alternatively be isolated from the high-quality top junctions by bifacial growth⁶⁷ or through the use of strain-balanced GaAsP/InGaAs bi-layers⁶⁸. The upright, metamorphic growth of lattice-mismatched top junctions on an active Ge junction (UMM) has also been demonstrated with high efficiency^{69,70}. Interestingly, very different 3J bandgap combinations used in the 3J IMM⁶⁰ and 3J UMM⁶⁹ resulted in very similar efficiencies as a result of the absorption gaps in the terrestrial spectra⁴⁶. More recently also 5-junction UMM solar cells were developed and used in concentrator photovoltaics⁷¹. Even higher quality low-bandgap junctions have been obtained by separately growing lattice-matched junctions on InP and wafer bonding with junctions grown on GaAs substrates for a 4J concentrator¹³ and 5J one-sun solar cell¹⁴. Space solar cells with AM0 efficiencies $> 32\%$ are available as 4J UMM and 5J IMM structures.

Ga_{1-x}In_xN alloys (that have been successful for light emitting devices using low-In content) have also been suggested for MJ solar cell materials because the alloy theoretically covers the full bandgap range³⁸. Some high bandgap GaInN solar cells have been demonstrated⁷², but the fabrication of the high-In GaInN alloys to capture the infrared portion of the solar spectrum has remained challenging. High-In GaInN alloys suffers from polarization charge⁷³, contact inversion layers⁷⁴, phase decomposition⁷⁵, and large lattice mismatch⁷⁶.

3.2. III-V Si Tandem

Silicon is a material which combines multiple benefits. It is earth abundant, almost 30% of the Earth's crust is formed from silicon. It can be purified to extremely high levels (typically less than 0.001% impurities in solar cell material) and grown into mono- or multi-crystalline ingots which are then diced and processed to solar cell devices absorbing sunlight between 300 to 1200 nm. The bandgap is close to ideal for a single-junction solar cell and has resulted in hero devices with up to 26.7 % conversion efficiency⁷⁷.

Mechanical strength and stability are further advantages and the cost of silicon solar cells has come down 82%^{*†} just between 2010 and 2020 mainly due to increased mass manufacturing and economies of scale. Fully processed devices are sold at approximately 0.12 US\$/Watt[‡] or 27 US\$/m² at an average solar-electric conversion efficiency of 22.5% AM1.5g. This is less than the price for many building materials like floor tiles. One could argue that the perfect solar cell material is already found but some limitations may still be overcome. Being an indirect semiconductor, silicon requires a certain thickness (typically 150 μm) to absorb sunlight, manufacturing processes are energy intensive and the conversion efficiency of a silicon single-junction solar cells is fundamentally limited to 29.5% by Auger-recombination^{78,79}. Auger recombination describes the three carrier energy transfer from a photogenerated carrier to an electron in the conduction band, and therefore becomes even more important at high concentration. This fundamental intrinsic recombination process determines the charge carrier lifetimes of ultra-pure silicon, different to direct semiconductors like GaAs or GaInP which are limited by radiative recombination.

The fundamental efficiency limit of silicon single-junction solar cells can be overcome by MJ devices as described above and such solar cells may still benefit from using silicon as the bottom junction and substrate material. This is attractive because most thin-film absorbers need a support as they are too thin to be self-sustained. And silicon with 1.1 eV bandgap is close to the optimum for dual-junction and triple-junction devices. The ideal bandgap energy for one additional absorber above Si is 1.7 eV and 2.0/1.5 eV in the case of two absorbers. The exact bandgaps depend on several factors such as transparency of the upper layers and the long-wavelength response of the silicon bottom solar cell. In fact, this has turned out to be one of the challenges in manufacturing tandem solar cells on silicon. The light-trapping features have to move from the front side to the rear of the wafer to allow the deposition of planar thin-film absorbers at the front⁸⁰⁻⁸². This can be done by implementing a pyramid texture⁸³, spheres⁸⁴, or nanostructure gratings⁸⁵ on the back side of the silicon wafer to increase the light path through the silicon and therefore enhancing absorption close to the indirect bandgap.

The most successful examples for Si-based tandem solar cells in terms of conversion efficiency are combinations of III-V compounds with Si (see example in Figure 7 right). GaAsP/Si tandem solar cells have reached AM1.5g conversion efficiencies of 23.4%

^{*†} Solar costs have fallen 82% since 2010, <https://www.pv-magazine.com/2020/06/03/solar-costs-have-fallen-82-since-2010/>

[‡] Prices according to <http://pvinsights.com/> (27.4.2021)

(monolithic, 2-terminals)^{86,87}, GaAs/Si up to 32.8 % (4-terminal)¹⁷, GaInP/Ga(In)As(P)/Si up to 35.9 % (2-terminal^{88,89}, and 4-terminal¹⁷). Some groups are growing the III-V layers directly on silicon which is a challenge due to the large difference in lattice constant of 3-4% and thermal expansion coefficient. Other groups have used wafer-bonding or gluing to make the connection, followed by a removal of the growth substrate. Of course, the latter can only be economically attractive if the growth substrate is removed with high yield and reused for further growth of III-V layers⁹⁰. Direct growth of the III-V absorbers onto the silicon using methods like metal-organic vapor phase epitaxy is challenging in terms of reaching low enough defect densities for highest efficiency devices but continuous progress is made in the field of metamorphic III-V growth on silicon^{91,92}. With further research this problem may be solved in the near future, making III-V direct growth the method of choice for realizing tandem solar cells on silicon which combine high performance, reliability and which can be manufactured at competitive costs.

3.3 Other MJ Architectures:

The ideal attribute for a MJ sub-cell material is one whose absorption threshold can be tuned over the solar spectrum, is stable, non-toxic, efficient and can be integrated into a tandem stack at low cost. Integration of sub-cells poses a particularly awkward challenge since sub-cells that can perform well in isolation can become impaired when integrated with additional sub-cells, either via impurity diffusion and/or excessive thermal budget. Mechanically stacking separate sub-cells in a multi-terminal device is one means by which incompatibilities can be overcome and serves as a useful proof of concept. For brevity we survey here only two terminal tandem solar cells fabricated from at least one novel material.

Metal halide perovskite materials are strong, direct-gap semiconductors with optical absorption, typically extinguishing sunlight in a layer of 200-400nm thick. Their absorption threshold can be tuned over a wide wavelength range owing to the wide range of alloy combinations with a ABX₃ structure; where A is an organic amine cation, B the metal cation and X the halide anion. Typical choices for A are methylammonium iodide Cs, B is commonly Pb and/or Sn and X is a halide, typically I and/or Br resulting in compounds such as FA_{0.75}CS_{0.25}Pb(I_{0.8}Br_{0.2})₃ a popular material for the top solar cell of a tandem. These materials offer a tantalizing and unprecedented range of photovoltaic solar cell absorber material combinations that can be prepared both via solution processing or vapour deposition⁹³. Several permutations of perovskite tandem have been attempted⁹⁴ and in particular the material system offers a range of opportunities

for achieving wide-gap absorbers that are well suited for many tandem solar cell applications⁹⁵.

3.3.1. Perovskite/Silicon tandems

The versatility of the perovskite material has made it a popular choice for a silicon tandem architecture as a top solar cell with potentially facile preparation illustrated in Figure 7(left). In this popular and fast-moving field, the aim is to establish a stable, wide-gap perovskite material^{95,96} that is compatible with a suitable silicon bottom solar cell⁹⁷. One of the outstanding challenges is to improve the voltage of the wide-gap perovskite top cell⁹⁵.

A 29.2% perovskite / silicon tandem solar cell was achieved by spin-coating a 1.68eV perovskite material onto a n-type heterojunction silicon solar cell with a textured rear-side only. Interconnection between the silicon and perovskite material was achieved using transparent, conducting metal oxide ITO layer rather than a tunnel junction⁴⁷. The device was stable under testing, retaining 95% of its initial efficiency after 300 hours of operation. An announcement of a 29.5% tandem solar cell was also recently made but no technical details are available at the time of writing[§]. A double textured perovskite / n-type heterojunction silicon tandem solar cell achieved a 25.2% power conversion efficiency where perovskite precursors were co-evaporated to form a conformal film over the textured silicon surface and interconnected using a nanocrystalline silicon tunnel junction⁹⁸. A similar result, 25.7% has also been achieved using a solution processed perovskite absorber and metal-oxide interconnection layer⁹⁹.

3.3.2. Perovskite/CIGS tandems

Copper Indium Gallium Selenide (CIGS) solar cells can also provide a convenient and commercially mature low-gap solar cell with strong absorption that with certain alloy fractions, deliver lower energy band-edge than silicon. The thin absorber enables thin, flexible solar cells to be made¹⁰⁰ and wide-gap perovskite materials offer an opportunity to augment the efficiency in a tandem configuration. Generally the film roughness of CIGS has complicated the fabrication of efficient tandem devices. The first reported Perovskite/CIGS tandem solar cell used a thick PEDOT:PSS layer achieving an efficiency of 10.9%¹⁰¹, later the CIGS was polished to yield a smooth surface and a much higher efficiency 22.4%¹⁰². More recently, self-assembled monolayers have been

§ <https://www.pv-tech.org/news/oxford-pv-pushes-tandem-shj-perovskite-cell-conversion-efficiency-to-record-29.52>

shown to form an effective interfacial layer between the perovskite and rough CIGS material resulting in a tandem efficiency of 24.2%^{103,104}.

3.3.3. Perovskite/Perovskite tandems

Absorption thresholds as low as 1.2eV can be achieved using mixed Pb-Sn perovskite materials¹⁰⁵ offering the opportunity for a perovskite/perovskite tandem solar cell. A 24.8% 1.77eV/1.22eV perovskite tandem solar cell has been achieved by paying particular attention to Sn oxidation and a low-optical loss tunnel junction that employed an ALD deposited SnO₂ layer interlayer between solar cells¹⁰⁶.

3.3.4. Organic tandems

The principle challenge for fabricating organic solar cells has been to find molecular absorber materials that operate efficiently in the infrared wavelength range¹⁰⁷. A 1.72eV PBDB-T:F-M/ 1.26eV PTB7-Th:O6T-4F:PCBM device achieved an efficiency of 17.3%¹⁰⁸. In comparison to the perovskite/perovskite tandem solar cell above, the principal loss in this organic tandem device is the low solar cell voltages obtained for each sub-cell, in addition to marginally impaired EQE and solar cell fill factor.

3.3.5. Chalcopyrite tandems

Fully inorganic thin-film tandems can be made using chalcopyrite materials, the well-established CIGS solar cell material is one alloy combination from a large array of the penternary $\text{Cu}(\text{In}_{1-x}\text{Ga}_x)(\text{S}_y\text{Se}_{1-y})_2$ material system that can span 1eV-2.43eV¹⁰⁹. A mechanical stack composed of a 1.48eV CdTe / 0.95eV CuInS₂ double junction achieved an efficiency of 15.3%¹¹⁰. A 1.68eV CuGaSe₂ / 1.1eV CuInGaSe₂ mechanically stacked device achieved an efficiency of 8.5%¹¹¹ while a 1.89eV GaInP/ 1.42eV GaAs / 1.20eV CIGSe mechanically stacked tandem has achieved 24.2%¹¹². Combining CIGS films into a monolithic tandem structure has proven difficult, owing to the complexity of forming the second junction without impairing the performance of the first. Alloying with Ag has provided a new dimension to tackle this problem since Ag alloys not only have marginally higher bandgaps, their lower melting point temperature helps reduce the thermal budget for forming the tandem solar cell and reduces compositional disorder¹¹³. Monolithic tandem efficiencies remain low (~3%)¹¹⁴ but the potential to exceed 25% with this approach exists if the difficulties associated with sub-cell integration can be overcome¹¹⁵. Monolithic chalcopyrite tandem devices on silicon have been attempted, a 1.65eV Cu₂ZnSnS₄ (CZTS) / 1.1eV Si tandem achieved an efficiency of 3.5%, likely limited by incomplete sulfurization and inadvertent silicon solar cell degradation¹¹⁶.

Higher efficiencies have been obtained for a 1.8 eV CdZnTe/ 1.1 eV Si tandem device that achieved 17 %¹¹⁷.

3.3.6. Antimony Chalcogenide tandems : Antimony selenosulfide $\text{Sb}_2(\text{S},\text{Se})_3$ forms 1D ribbons¹¹⁸ and by varying the Se/S atomic ratio offers an adjustable absorption threshold from 1.7eV to 1.1eV. To date a 10 % efficient single-junction solar cell has been demonstrated¹¹⁹ and a proof-of-concept 1.74eV Sb_2S / 1.22eV SbSe achieved an efficiency of 7.9 %¹²⁰.

3.3.7. Organic-Silicon tandems: Organic absorber materials are well suited for absorbing visible wavelengths and can therefore form the high bandgap junction in a hybrid organic-silicon device. A dye sensitized solar cell was partnered with a silicon solar cell to form a 1.8eV dye / 1.1eV Si mechanical stack tandem cell with an efficiency of 14.7%¹²¹. The convention for interconnection of a tandem solar cell is a series connected stack, but this is only one means by which multiple absorbers can be arranged, several other permutations are possible in a combination of series and/or parallel connection¹²². Specifically the combination of a wide-gap solar cell in parallel with two lower gap solar cells has the merit of lower sensitivity to variation in the incident solar spectrum¹²³ has been demonstrated as a so-called ‘voltage matched’ tandem whereby a pentacene layer absorbs photons at energies above 1.8eV that undergo singlet fission to produce two electron hole pairs¹²⁴. A fully parallel singlet fission device has also been demonstrated¹²⁵ in addition to a conventional series connected tandem¹²⁶.

All the results reported in section 3.3 are derived from small area cells, most smaller than 1cm^2 . For all these approaches to achieve their practical potential, high efficiency will need to be maintained over large areas. III-V multi-junction solar cells are manufactured on 6” wafers and subsequently interconnected in series to form a module. The promise of thin-film tandem cells to which all but the silicon based tandems aspire, is to expand the substrate size significantly, ideally coating an entire sheet of module glass. One of the few tandem technologies that achieved this on a manufacturable scale was the micromorph tandem from Oerlikon, where an amorphous silicon a-Si / microcrystalline silicon $\mu\text{c-Si}$ tandem configuration was manufactured over an area of 1.4m^2 delivering an initial 11 % module efficiency. The micromorph technology was rendered obsolete when the costs of the more efficient c-Si modules dropped significantly below that of the micromorph tandem.

IV. PERSPECTIVE

4.1. Efficiency Improvement and Cost Reduction Potential of MJ Solar Cells

4.1.1 High efficiency potential of MJ solar cells

Analysis of the performance of MJ solar cells has been carried out by several researchers¹²⁷⁻¹³⁵. Figure 8 shows the calculated efficiency potential of MJ solar cells under 1-sun and concentration conditions as a function of number of junctions reported by some groups^{134,135}. As shown in Fig. 8, increasing the number of junctions, that is, increasing the number of sub-cells in MJ solar cells is effective to increase conversion efficiency of these solar cells. Single-junction and 3-junction solar cells have potential efficiencies of more than 50% and 60% under 1-sun illumination in the ideal detail-balance limit, respectively. However, only 80% and 85% of ideal efficiencies are pointed out to be achievable in practical devices¹³⁴. Figure 9 shows calculated efficiencies of III-V compound MJ solar cells under 1-sun conditions as a function of the number of junctions and external radiative efficiency (ERE) in comparison with efficiency data (best laboratory efficiencies) reported in references^{11-15,136}. Ideal efficiencies of MJ solar cells can be calculated by estimating the short-circuit current density of sub-cells from the standard solar spectrum, by considering only radiative recombination loss (0.26V loss for Si and 0.28V loss for GaAs compared to E_g/q ; E_g is the bandgap energies of sub-cells, q is the electronic charge)¹³⁷ in sub-cells and resistance loss (only 1%) and the similar potential efficiencies of MJ solar cells with reported values^{134,135} can be estimated. In the more realistic case, the combination of sub-cells is often selected by considering lattice matching of sub-cell lattice constants and ease of sub-cell material growth. Mismatching of short-circuit current densities of sub-cells by considering bandgap energy and layer thickness of sub-cells, non-radiative recombination and resistance losses are further considered. In the case of MJ tandem solar cells, we define average ERE (ERE_{ave}) by using average open-circuit voltage V_{oc} loss:

$$\Sigma(V_{oc,n} - V_{oc,rad,n})/n = (kT/q)\ln(ERE_{ave}), (1)$$

where V_{oc} is the measured open-circuit voltage, $V_{oc,rad}$ is the radiative open-circuit voltage, n is the number of junctions, k is the Boltzmann constant and T is the absolute temperature. The resistance loss of a solar cell is estimated solely from the measured fill factor. In Fig. 9, best efficiencies^{11-15,136} of III-V compound MJ solar cells are also plotted. 2, 3, 4, 5 and 6-junction solar cells have potential efficiencies of 36.6%, 44.0%, 48.8%, 50.4% and 51.4%, respectively, as shown in Fig. 9. Table 1 shows major losses, their origins of III-V compound MJ solar cells and key technologies for improving efficiency. As shown in Table 1, further development of MJ solar cells is necessary in order to realize optimum

efficiencies.

Other MJ solar cells composed of II-VI, chalcopyrite, kesterite compound and perovskite solar cells are thought to have the similar potential as III-V compound MJ solar cells. In order to realize high efficiency MJ solar cells using these materials, reducing non-radiative recombination and resistance losses by learning from progress in III-V compound MJ solar cells is necessary.

4.1.2. Cost analysis of MJ solar cells

The allowable cost per unit area of solar cell modules depends strongly on module efficiency^{138,139}. For example, a 30%- efficient solar cell costing 3.5 times as much as a 15%-efficient solar cell of the same area will yield equivalent overall photovoltaic system costs¹³⁸ due to the balance of system costs. Therefore high-efficiency solar cells will have a substantial economic advantage over low-efficiency solar cells providing the cost of fabricating them is low enough. Additionally, efficiency improves the environmental impact of photovoltaic modules as less material is needed. For space applications, high-efficiency solar cells have significant payload advantages. Although the III-V MJ solar cells have demonstrated an extremely high conversion efficiency with up to 39.2%¹⁵, further cost reduction is still necessary to access terrestrial photovoltaic markets.

Figure 10 shows a comparison of expected module costs as a function of module production volume for III-V tandem cells with/without high-speed deposition, III-V/Si tandem devices, and concentrators, reported by the authors¹³⁹ and cost analytical results for rapid deposition (HVPE; Hydride Vapor Phase Epitaxy)¹⁴⁰ and Si tandem¹⁷ reported by NREL. Therefore, ways for module cost reduction are reduction in film thickness, a high growth rate of the III-V layers, reuse of substrates, concentration of light, use of Si as substrate material and bottom cell, and an increase in module production volume as shown in Fig. 10. The results suggest that there are ways to realize costs of less than \$1/W for III-V compound MJ solar cell modules by scaling up production volume to 100 MW/year with a high-speed growth method, or with Si-based tandem solar cells. Many of these technologies are current fields of research.

Cost analysis of perovskite and perovskite/Si tandem solar cells has been reported^{141,142}. About \$0.5/W, comparable with crystalline Si and CdTe solar cell modules, was estimated as a manufacturing cost of perovskite and perovskite/Si tandem solar cells. One highly uncertain aspect of the module cost for tandem solar cells is the level of encapsulation that will be required to maintain tolerable module performance over the working lifetime, usually 20 years. For a silicon tandem, the underlying silicon cell degradation is generally low (less than 0.5%/year) with standard encapsulation methods

but it is not known what level of encapsulation will be required to deliver long-term efficient operation from perovskite on silicon tandem solar cells. What has been established is that the degradation rate of the top cell should be below 0.9%/year for a silicon tandem to remain financially viable¹⁴³.

4.2. Perspective for MJ Solar Cells

Of all the so-called third generation solar cell strategies⁸, only MJ designs have been successful to surpass the detailed-balance limit of single-junction solar cells. Table 2 shows potential efficiencies of 3rd generation solar cells^{8,141-143}. Demonstrated efficiencies of III-V MJ solar cells at one-sun are nearing 40% and under concentration are approaching 50%. Most of the physics are understood, but sophisticated engineering and high-quality materials are absolutely required. While low-cost solar cell materials are desirable for tandems solar cells, only high-voltage junctions, as quantified by the ERE^{26,147}, with well-chosen bandgaps matched to the application spectra are useful to surpass the efficiency of single-junction silicon. Quantification of spectral efficiency¹⁴⁵ is a convenient metric to judge how to choose tandem partners, and more comprehensive multi-junction models are also available¹⁴⁹. The challenge for low-cost tandem materials is to bring the best together in a way that preserves the high-quality junctions. This is already achieved in high efficiency III-V multi-junction devices but here lower processing costs are needed without compromising on required quality for flat-plate areas. Alternatively, renewed development of robust and economical terrestrial concentrator systems could result in high demand for the most efficient multi-junction solar cells possible. The continued development of multi-junction solar strategies through multiple pathways and a sufficiently large market is likely to bring the technology closer to the economics of single-junction silicon and to provide clean, economical, and efficient energy especially for area constrained applications.

4.3. Perspective for Si-based Tandem Solar Cells

It can be expected that silicon-based tandem solar cells will receive further growing attention by the photovoltaic industry as efficiencies for single-junction solar cells reach a plateau and many of the opportunities for cost reduction have already been implemented. Innovation will most likely come from more efficient devices and it has already been shown that Si-based tandem solar cells can reach nearly 36% conversion efficiency in 2- or 4-terminal configuration. This proves that Si-based tandem cells can get close to the

best triple-junction solar cells ever reported in literature and radiative efficiency limits are even as high as 45.2%, 49.6%, 52.2% for 2-junction, 3-junction and 4-junction cells¹⁵⁰ leaving sufficient room for further development. Using combinations of silicon and III-V materials allows the PV industry to keep many established processes which have already been scaled to large volumes. But at the same time new processes must be implemented and scientists will continue to have different opinions on which technologies are favorable for silicon based tandem solar cells. The market will accept all those solutions which provide sufficiently high efficiencies combined with economically attractive production processes. We believe that III-V/Silicon tandem solar cells must have a significant efficiency advantage compared to conventional silicon single-junction devices because they will be more expensive in the beginning. The exact efficiency number for market entry may be debatable but probably it is on the order of 30% (AM1.5g) or more. Cost will come down with market size as indicated in Figure 10 but for this to materialize, entry markets must be found. An example is electric cars where the high performance of the solar cells directly translates into longer driving distances before re-charging of the battery. Such conveniences often convince companies and customers to pay a premium price.

The simplest solution to realize a III-V/silicon tandem solar cell is a 2-terminal device where the GaAsP top cell is grown directly onto a silicon bottom junction. The silicon junction may be formed by diffusion or implantation of P, GaP serves as front surface field and the rear side may be formed by a combination of a SiO_x passivation with a nanostructured grating for light diffraction. Aluminum can be sputtered and point contacts formed by laser firing. Such a device would fulfill the requirements of manufacturability and low cost but the performance is today still falling behind a good silicon single-junction solar cell. This may certainly change in the next coming years as defects in the III-V epitaxial layers are better controlled and issues with the GaP passivation to silicon solved. But it may also be necessary to develop more complex device architectures implementing III-V layer transfer from a GaAs substrate, tunnel oxide passivation for silicon, 4-terminal architectures or additional junctions. Some of these approaches already reached up to 36% efficiency but suffer in terms of manufacturability and cost. Finding the path towards a III-V/silicon tandem product will remain subject to continued discussion in coming years. But we can be confident that once performance and cost are shown to be economically attractive, nothing will stop this technology from growing to large scales. Reliability should be comparable to silicon solar cells and there seems to be no restriction to scale manufacturing to the GW level.

V. CONCLUSION

In order to realize a society based on renewable energy, solar cells with highest efficiency are very attractive because they reduce the required system area and need for materials. As single-junction solar cells are limited to 30-32% conversion efficiency under 1-sun, MJ or tandem solar cells are expected to contribute to higher performances. The concept of MJ solar cells was first and most successfully implemented using III–V compound semiconductors and such products have already become the standard technology in space. III-V MJ solar cells need to further improve their conversion efficiency and reduce their cost to achieve widespread terrestrial deployment. At the same time perovskite materials have appeared as an alternative solution to form MJ devices but here reliability, module integration and large volume manufacturing are still subject of current research and development.

In this paper, we provide perspectives for MJ solar cells from the viewpoints of efficiency and low-cost potential based on scientific and technological arguments and possible market applications. 2, 3, 4, 5 and 6-junction solar cells have potential efficiencies of 36.6%, 44.0%, 48.8%, 50.4% and 51.4% under 1-sun, respectively. For realizing higher efficiency MJ solar cells, we showed the importance of improving external radiative efficiency of the solar cell materials or in other words improving material quality, decreasing defect density in the bulk and at interfaces. Further decreasing resistance losses and applying light management for better absorption or photon recycling are important features to accomplish. There is a wide range of technological options under development which will lead to further efficiency improvement in the future where most developments target III-V MJ solar cells, III-V/Si, II-VI/Si, Perovskite/Si tandem solar cells. The potential for < 1 \$/W MJ solar cell modules exists for III-V based devices if new technologies such as high-speed deposition, Si-based tandem solar cells or the use of concentration are matured and realized with sufficient efficiency and manufacturability. Once the technology meets some terrestrial markets, cost reduction will happen, driven by the increase in production volumes. This is well known from the history of the silicon photovoltaic industry. So the main question is how to approach and enter terrestrial markets with volumes on the order of several hundred MW/year. Besides, the III-Vs, other materials may have advantages in terms of production cost and they can enter the field of multi-junction technology quickly once the materials show high external radiative efficiency and reliability. II-VI/Si, chalcopyrite/Si and especially perovskite/Si tandem solar cells are developing quickly and are expected to play an important role as high-efficiency and low-cost solar cells in the future. As ideal

bandgap combinations for highest efficiency MJ solar cells are often found in lattice-mismatched systems, efficiency improvements by reduction in bulk recombination based on further understanding of non-radiative recombination are necessary. Reduction in surface and interface recombination, efficient optical coupling and low loss electrical interconnection of sub-cells, and effective photon recycling of bottom solar cell are also key elements for high-efficiency MJ solar cells. At this point nobody can predict which concept will be most successful, but we believe clearly that at least one multi-junction solar cell technology with efficiency beyond the limits of silicon will appear as a major player in the photovoltaic market. The question is only which materials will take the lead, and when this will happen.

DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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Table 1. Major losses, their origins of III-V compound MJ solar cells and key technologies for improving efficiency. (Reproduced with permission from Ref. 2 [Wiley], [2015]).

Losses	Origins	Technologies for improving
Bulk recombination loss	Non radiative recombination centers (impurities, dislocations, other defects)	High quality epitaxial growth Reducing thermal stress Reduction in density of defects
Surface recombination loss	Surface states	Surface passivation Heterointerface layer Double hetero structure (sandwiched with higher bandgap barrier layer)
Interface Recombination loss	Interface states Lattice mismatching defects	Lattice matching Inverted epitaxial growth Back surface field layer Double hetero structure
Voltage loss	Non radiative recombination Shunt resistance	Reduction in density of defects Thin absorber layers
Resistance loss	Series resistance Shunt resistance Loss of sub-cell interconnection	Reduction in contact resistance Reduction in leakage current, Surface, interface passivation Reduction in sub-cell interconnection loss
Optical loss	Reflection loss Insufficient absorption	Anti-reflection coating, texture Back reflector, Photon recycling
Insufficient-energy photon loss	Spectral mismatching	Selection of sub-cell materials

Table 2. Potential 1-sun AM1.5g efficiencies of 3rd generation solar cells.

Concept	Potential Efficiency	Achieved Efficiency
Hot carrier solar cells	68%	11.1% @50,000-suns [ref. 144]
Tandem (Multi-junction) solar cells ($n \rightarrow \infty$)	68%	39.2% (n=6) [ref.15] 47.1% @143-suns (n=6) [ref.15]
Thermophotovoltaic solar cells	54%	29.1% at emitter temperature of 1207 °C [ref.145]
Tandem solar cells (n=3)	49%	37.9% [ref.12] 44.4% @300-suns [ref.12]
Impurity band solar cells (Quantum dot solar cells)	48%	18.7% [ref.146]
Tandem solar cells (n=2)	43%	32.8% [ref.1] 35.5% @38-suns [ref.1]
Single-junction solar cells	31%	29.1% [ref.1] 30.5% @258-suns [ref.1]

Figure captions

FIG.1. Calculated and obtained efficiencies of single-junction single-crystalline and polycrystalline solar cells. ERE shows external radiation efficiency expressed by eq. (1) and the solar cells with the higher ERE value show the less of non-radiative recombination loss. Efficiency values of cells with an area of 1cm^2 or larger area was plotted in the figure. (Reproduced with permission from Ref. 2 [Wiley], [2015] and updated.).

FIG. 2. Loss processes and power out in an unconstrained, MJ device under one sun illumination (6000K Blackbody) are shown. All incident solar radiation is accounted for. Optimal bandgaps are used in each case. All mechanisms are shown to be dependent on the number of junctions. (Adapted with permission from Ref.6 [Wiley], [2011]).

FIG. 3. Principle of wide photoresponse by using a MJ solar cell, for the case of an GaInP /GaAs/Ge triple-junction solar cell. (Adapted with permission from Ref. 7 [Toyota Tech. Inst.], [2020]).

FIG. 4. Chronological improvements in conversion efficiencies of concentrator MJ and MJ solar cells in comparison with those of crystalline Si, GaAs, CIGS and perovskite single-junction solar cells. (Reproduced with permission from Ref. 7 [Toyota Tech. Inst.], [2020] and updated.).

FIG. 5. Fundamental processes in an ideal solar cell : Schematic of a single-junction solar cell of bandgap E_g showing four fundamental processes: (1) absorption of light above the bandgap energy (E_g), (2) transmission of light below the bandgap energy, (3) thermalisation of excess energy, (4) radiative recombination.

FIG. 6 Bandgap vs lattice-constant of III-V semiconductor alloys. Various MJ solar cell combinations are also shown with demonstrated AM1.5 global one-sun and AM1.5 direct concentrator efficiencies. Most successful III-V MJ solar cell designs use GaAs and

GaInP junctions.

FIG. 7. Examples for the layer structure of a perovskite/Si dual-junction solar cell structure with pyramids on the rear (left) and a GaInP/GaAs/Si triple-junction solar cell which uses a nanostructured resists grating as diffusor on the back (right).

FIG. 8 Calculated efficiencies of MJ solar cells under 1-sun and concentration conditions as a function of number of junctions. Calculated efficiency values are taken from ref.134 and 135.

FIG. 9. Calculated efficiencies of III-V compound MJ solar cells under 1-sun conditions as a function of number of junctions and external radiative efficiency (ERE) in comparison with efficiency data (best laboratory efficiencies) reported in references.

FIG. 10. A comparison of module cost as a function of module production volume for III-V tandem, high-speed deposition, Si tandem, and concentrator, reported by the authors and cost analytical results for rapid deposition (HVPE; Hydride Vapor Phase Epitaxy) and Si tandem reported by NREL. (Reproduced with permission from Ref. 139 [Elsevier], [1994] and updated.).