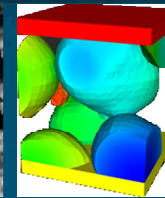
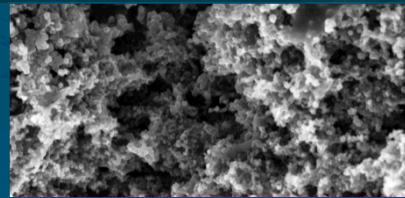
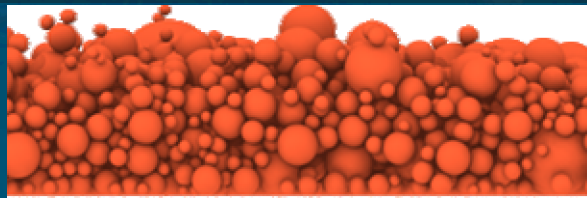
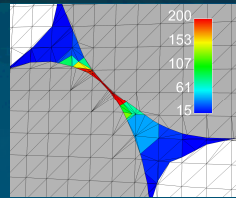
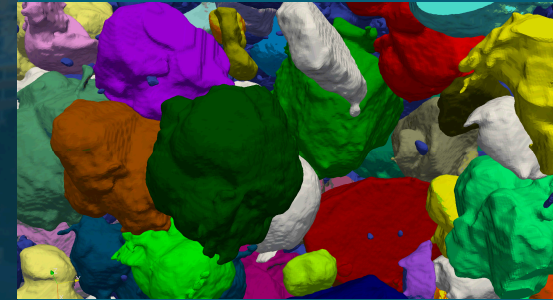


Modeling Mechano-Electrochemical Phenomena from the Mesoscale to the Macroscale



- Srikanth Allu (ORNL)
- Victor E. Brunini
- Mark E. Ferraro
- Jeremy B. Lechman
- Aashutosh N. Mistry (Purdue)
- Partha P. Mukherjee (Purdue)
- David R. Noble
- Chance Norris (Purdue)
- Ishan Srivastava
- John Turner (ORNL)
- Bradley L. Trembacki

PRESENTED BY

Scott A. Roberts, Ph.D.

Princeton University Seminar
July 23, 2018



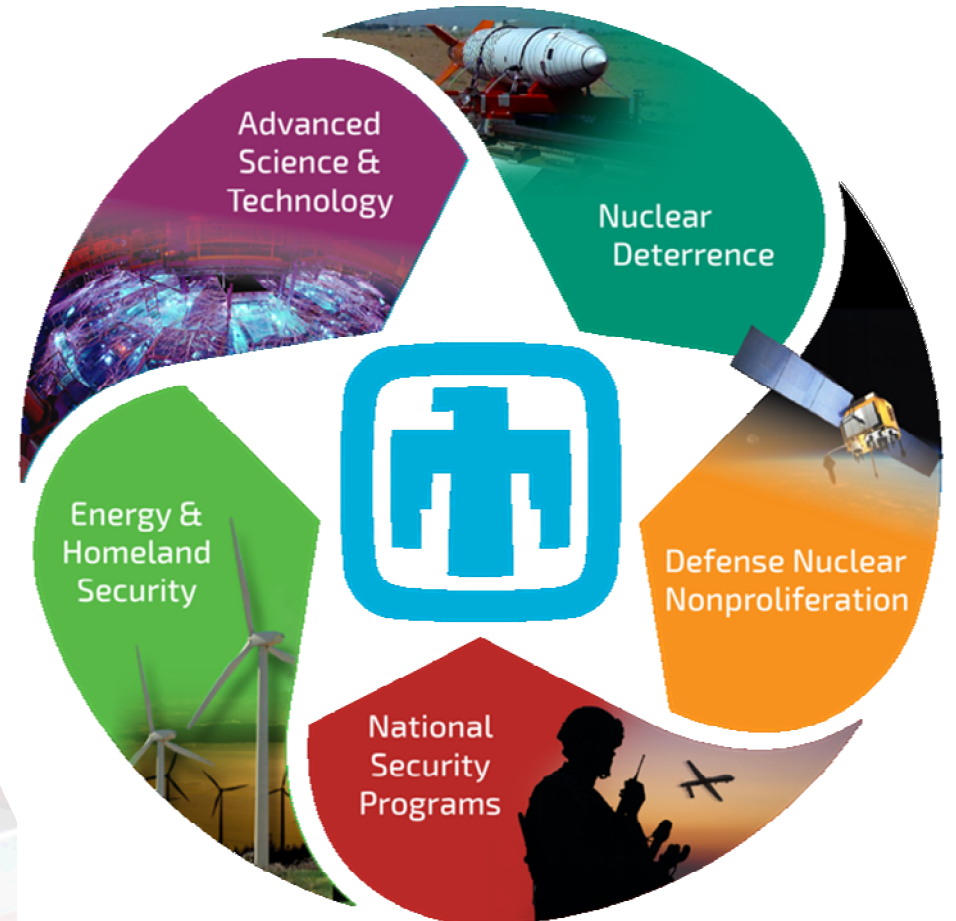
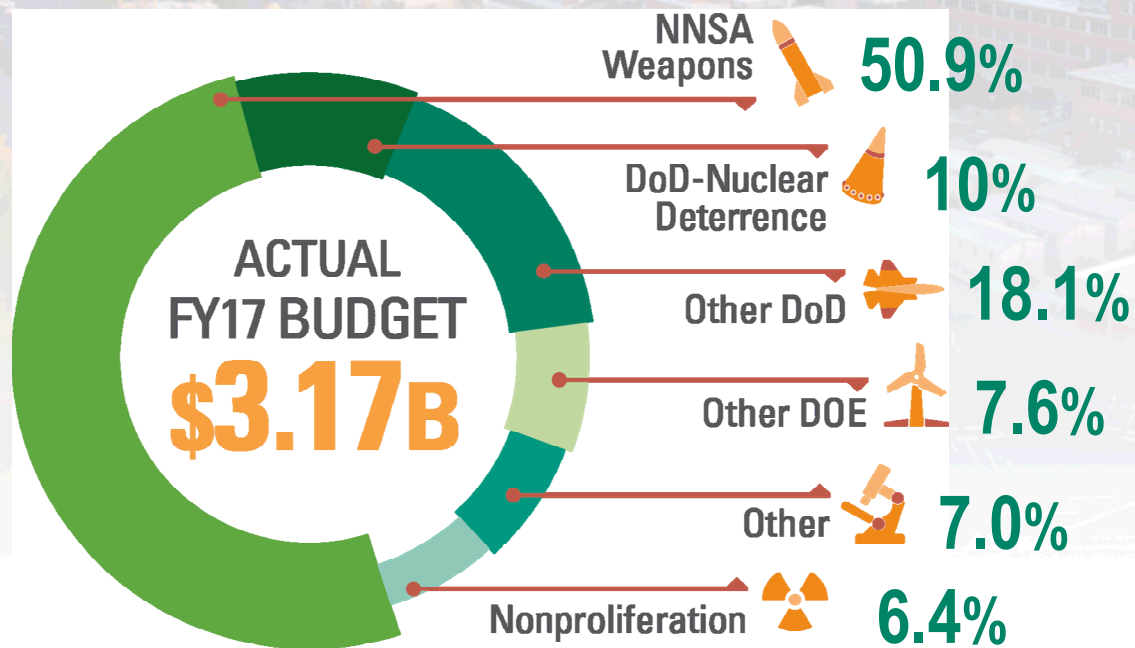
Sandia National Laboratories is a multimission laboratory managed and operated by National Technology and Engineering Solutions of Sandia LLC, a wholly owned subsidiary of Honeywell International Inc. for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

UNCLASSIFIED
RELEASE PENDING REVIEW & APPROVAL

2 Sandia National Laboratories

Sandia is the nation's premiere engineering DOE laboratory

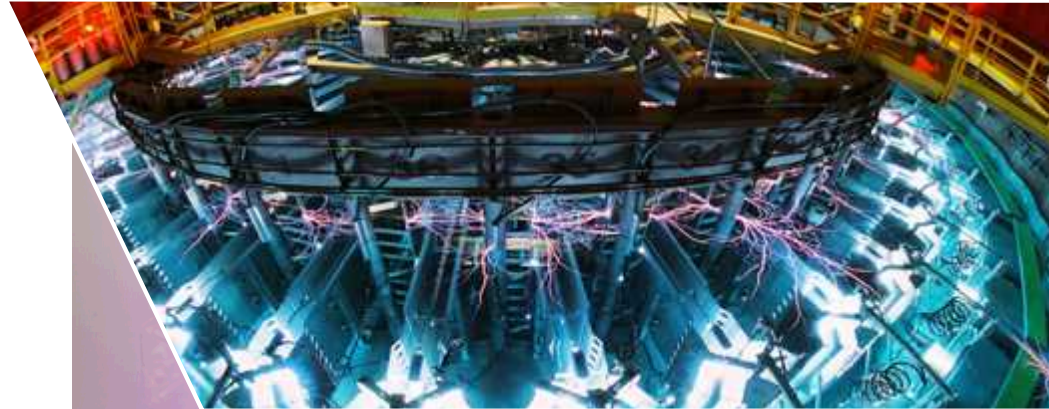
- Established in 1949 from Los Alamos Z Division
- Primary locations: Albuquerque, NM; Livermore, CA
- 12,200 employees (52% R&D)



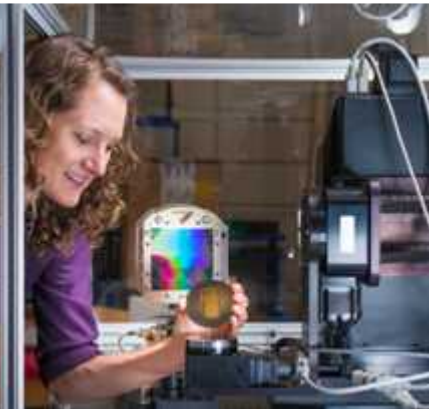
Advanced Science & Technology

Research & Technology in support of the central role in mission delivery

Nanodevices & Microsystems



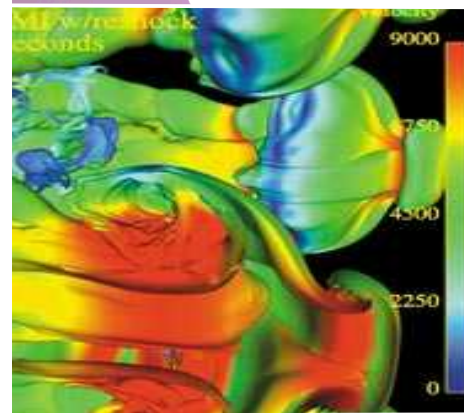
Radiation Effects & High Energy Density Science



Materials Science



Computing & Information Technology



Engineering Science



Geoscience

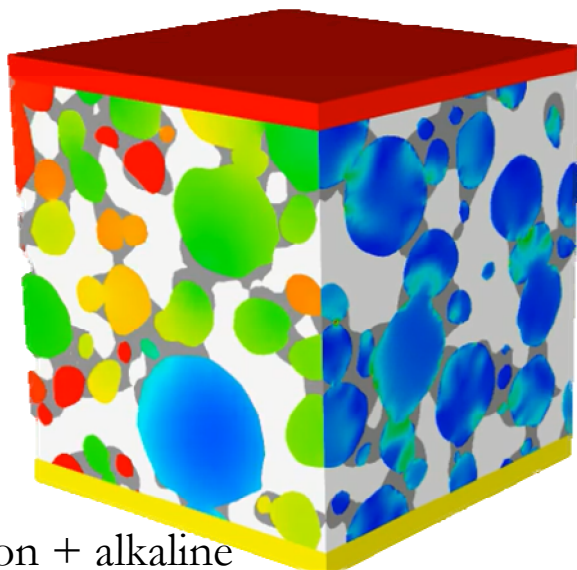


Bioscience

4 Roberts research efforts

Multi-physics, multi-scale applied modeling & simulation

- Lead teams comprised of 14 PhD staff members, 3 post-docs, 2 graduate students, and 4 interns
- \$2.9M in FY18 research funding



Li-ion + alkaline
battery mesoscale

EM railgun launcher performance

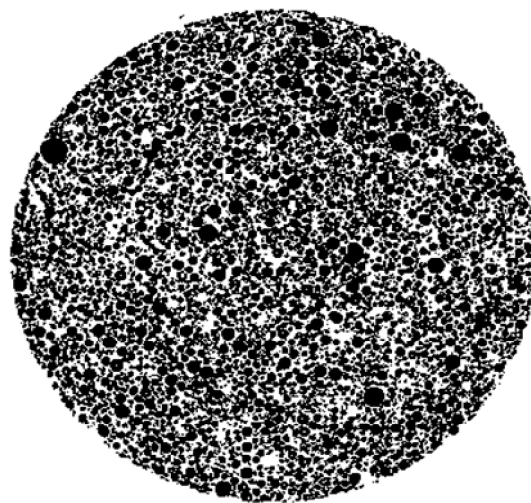
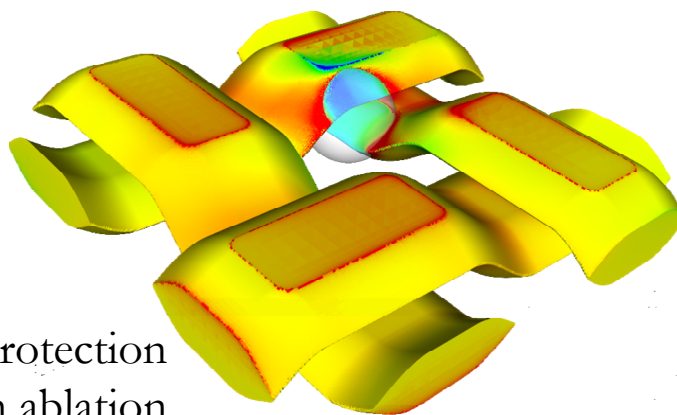
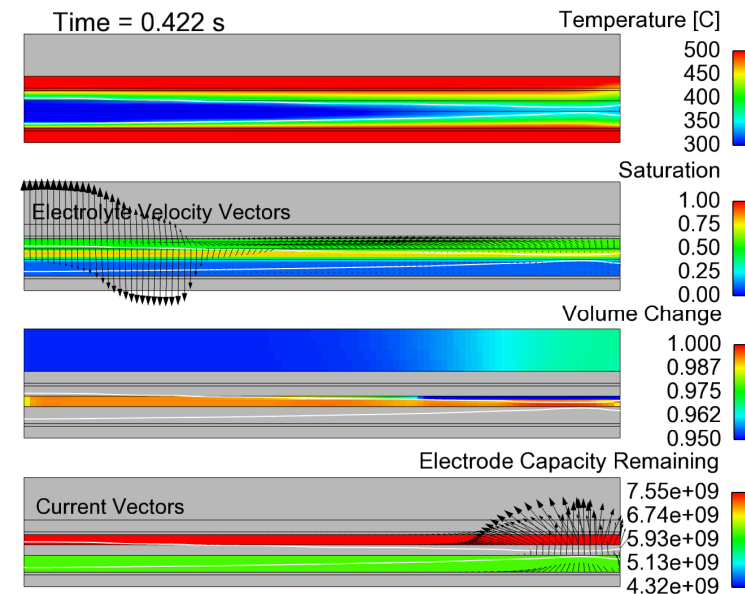


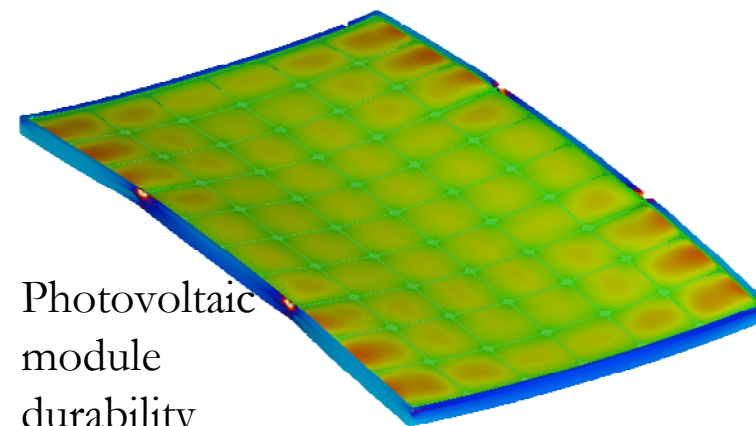
Image-to-mesh with ML/UQ



Thermal protection
system ablation



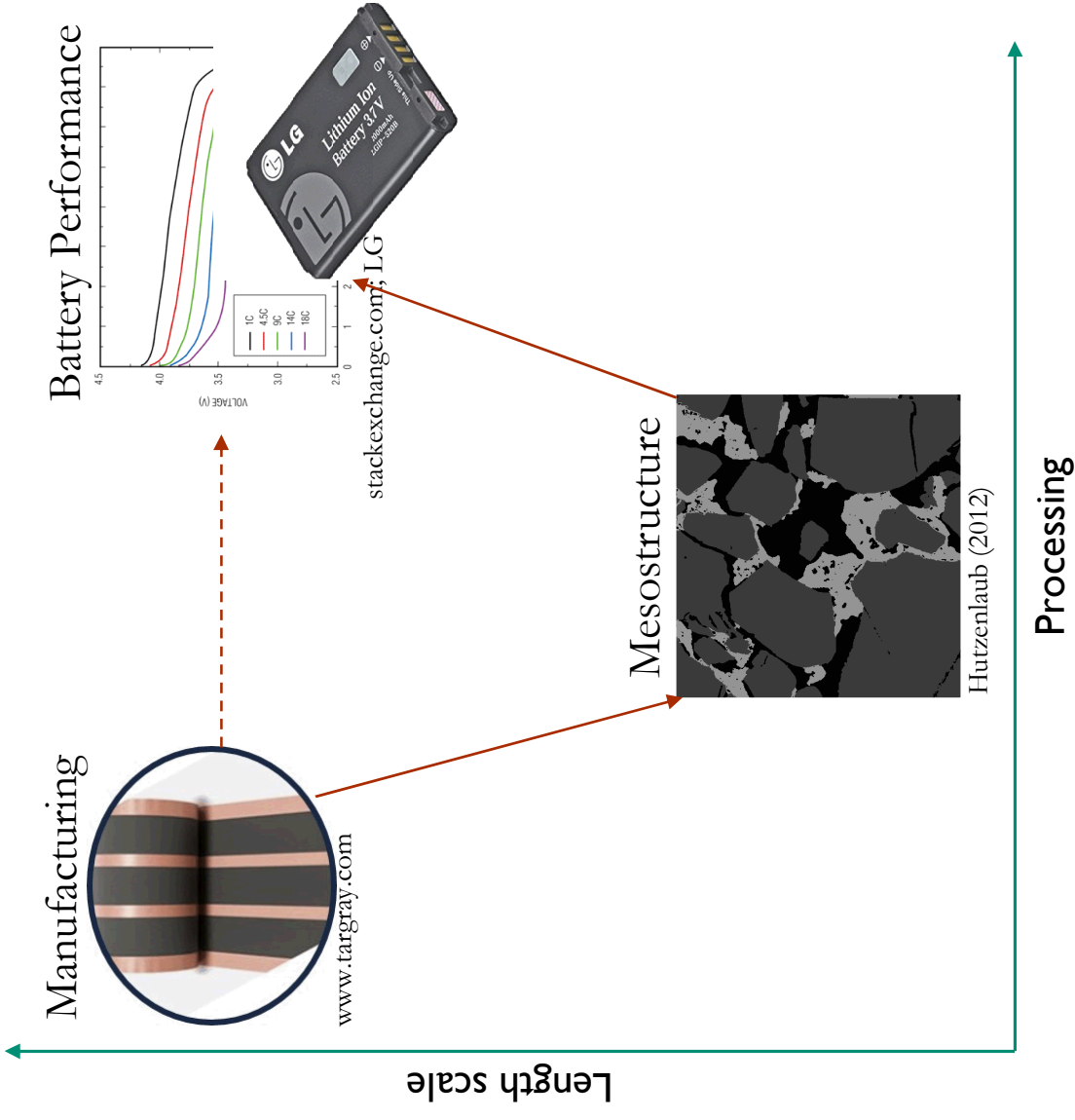
Thermal batteries



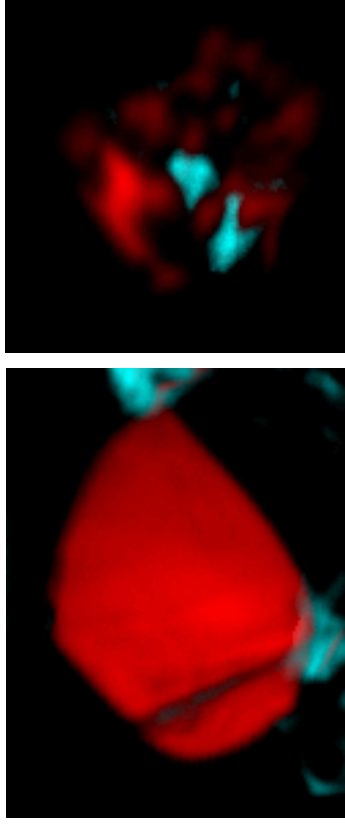
Photovoltaic
module
durability

Motivation

5



Lithiation-Induced Fracture



LiCoO₂ STXM, Farid El Gabaly Marquez (Sandia)

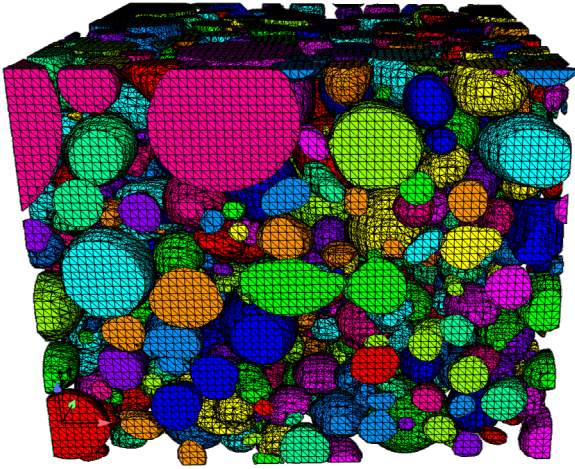
Mechanical Abuse → Electrochemistry / Mesostructure



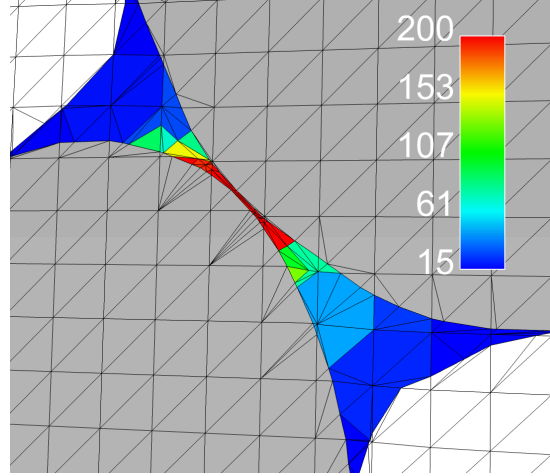
NMC/graphite pouch, H. Wang (ORNL)

Cannarella/Arnold (2015)

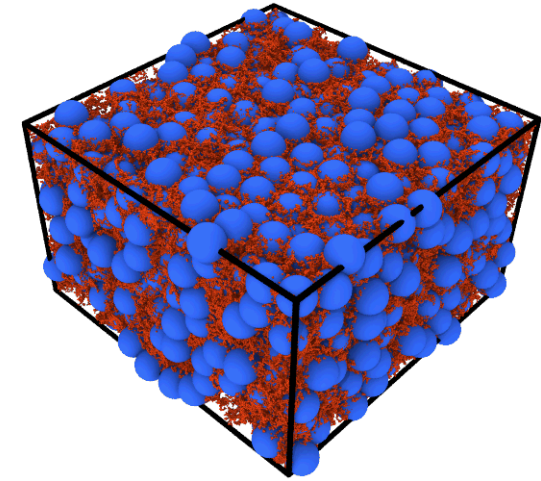
Coupled electrochemical-mechanical effects at mesoscale connect battery manufacturing and performance



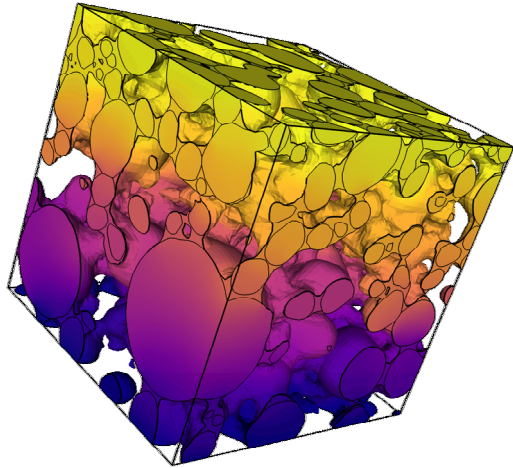
Computational representation of electrode mesostructures



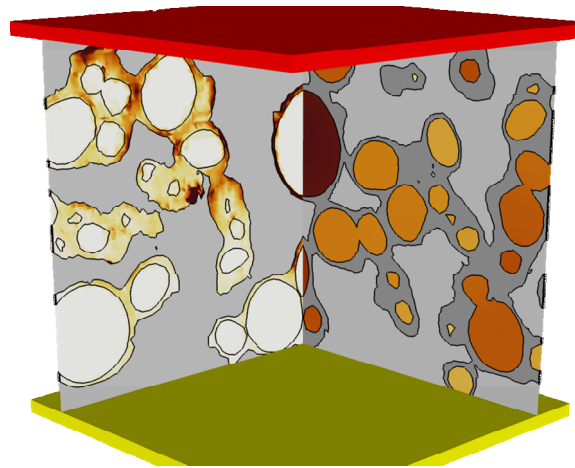
Representation and role of conductive binder morphology



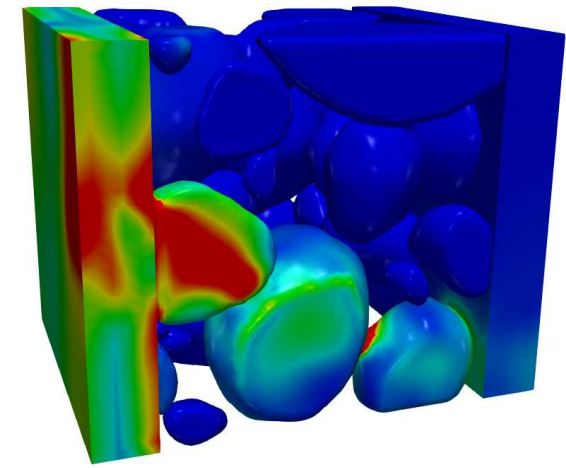
Discrete element method mesostructure generation



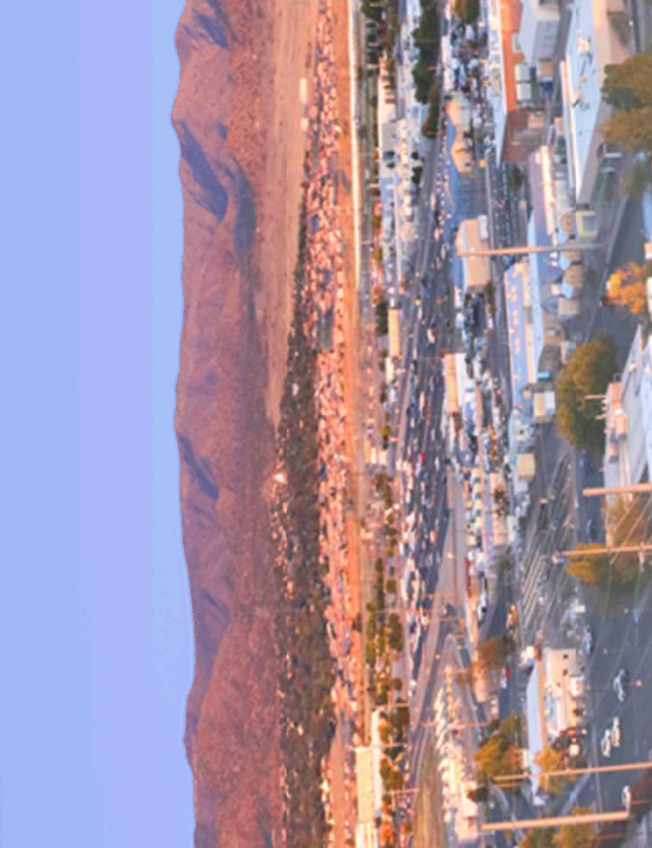
NMC cathode effective property prediction and upscaling



Electrochemical-mechanical discharge sims. of NMC half-cells

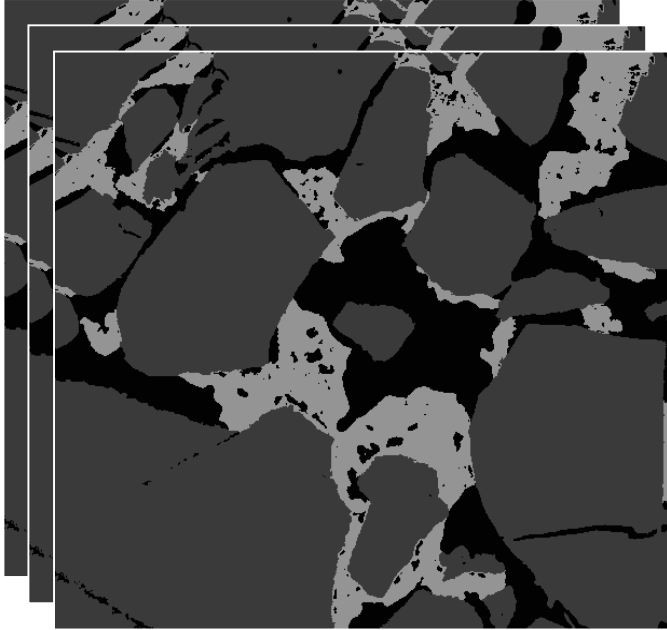


Future directions in electrode mesoscale modeling

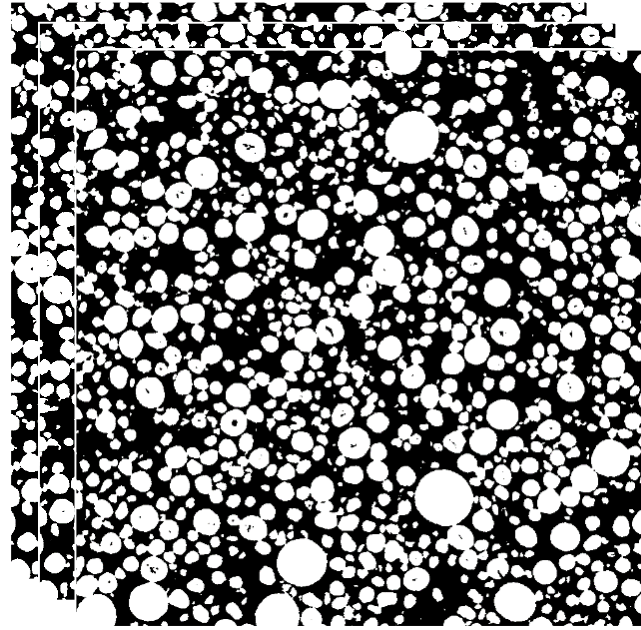


Computational representation of electrode mesostructures

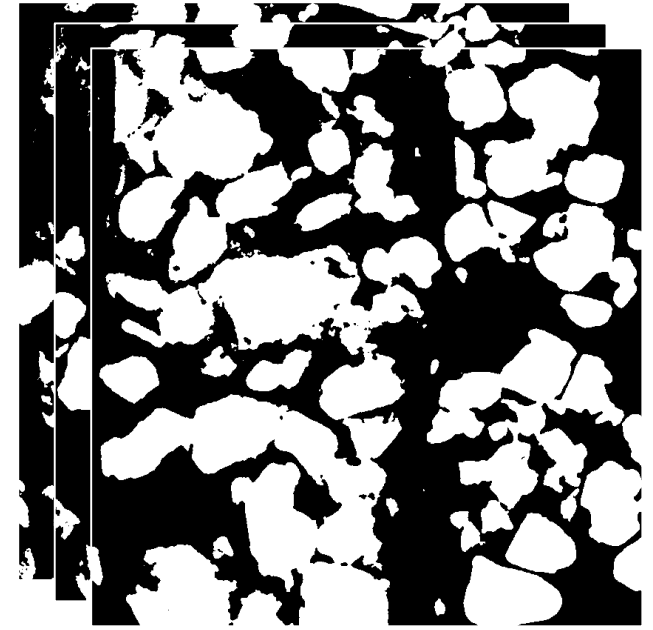




LCO with binder from FIB/SEM,
35 nm resolution,
20 μm domain.
Hutzenlaub (2012)



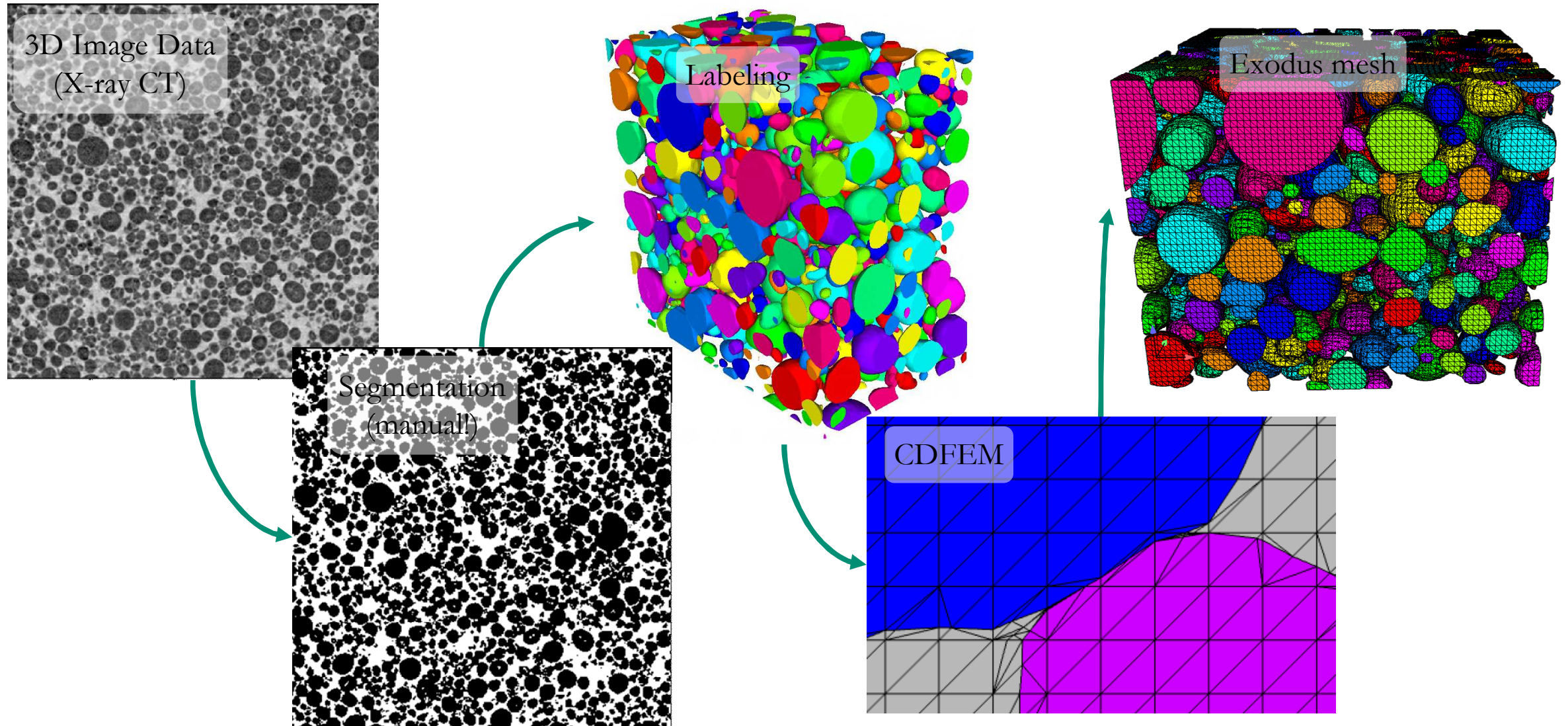
NMC from XRCT,
370 nm resolution,
757 μm domain.
Ebner (2013)



LCO from XRCT,
64 nm resolution,
22 μm domain.
Yan (2012)

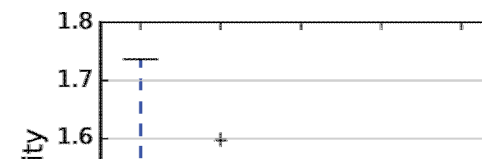
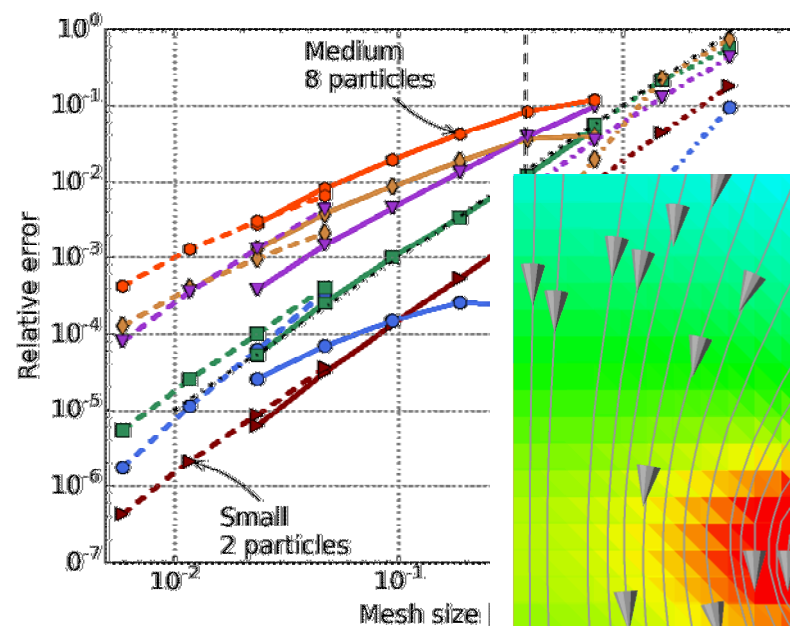
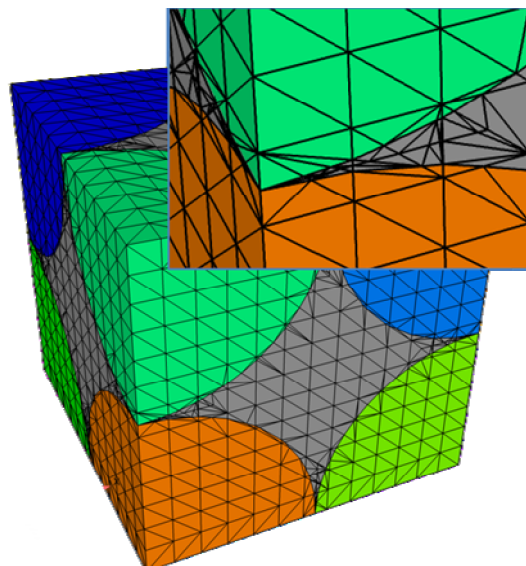
Imaging reveals complex networks; binder can be difficult to detect at scale

9 Mesoscale geometry from CT data using CDFEM

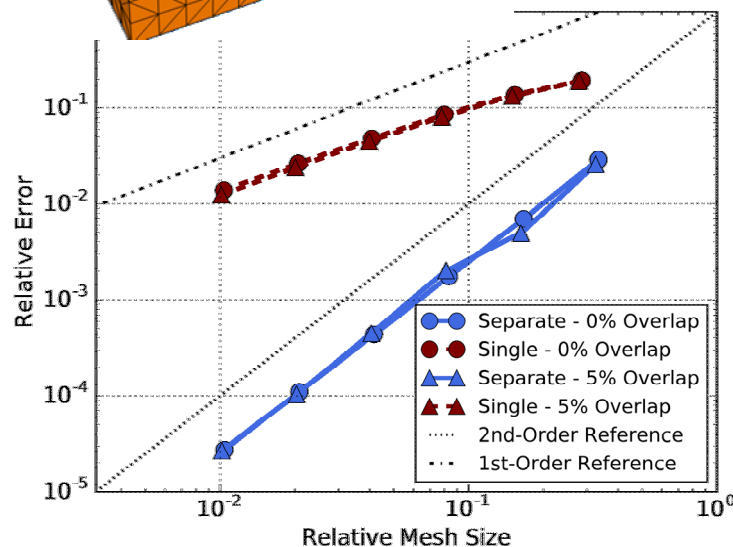


Detailed 3D reconstruction and image processing necessary to get usable mesostructure data

Solution verification for credible mesostructure simulations

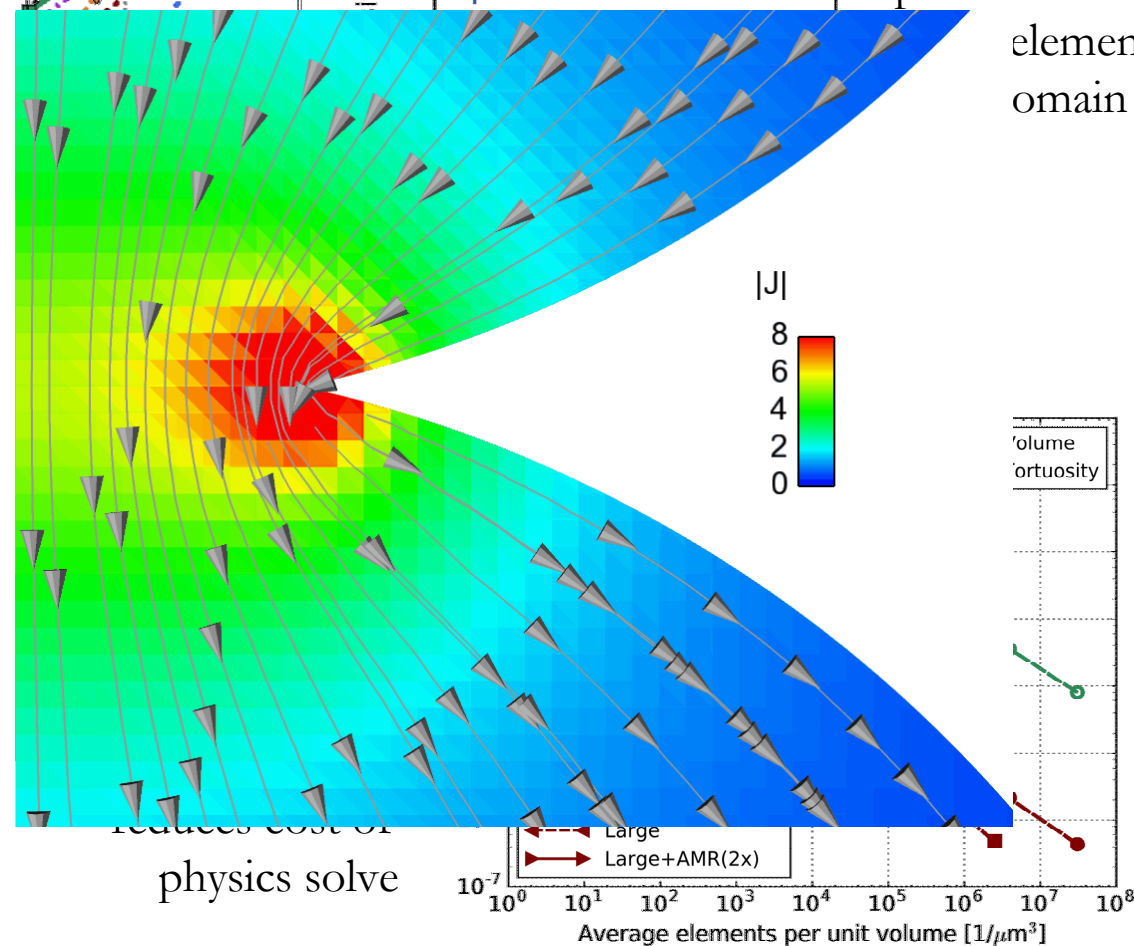


Quantify
representative
element
domain

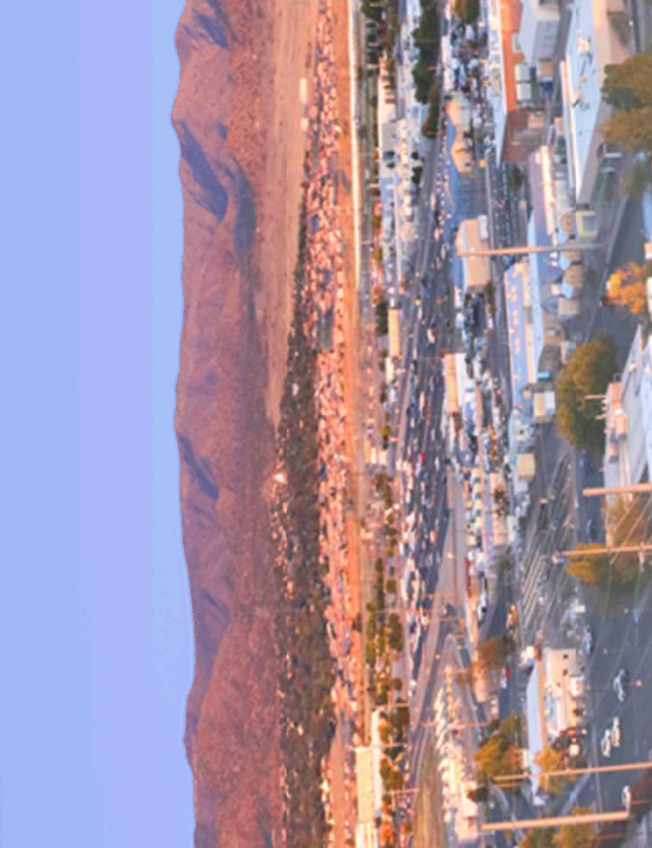


Understand mesh resolution
for accurate prediction

Separate particle
representations
required for
optimal
convergence



Solution verification establishes simulation correctness and domain/mesh size requirements



Representation and role of conductive binder morphology



What about the conductive binder?

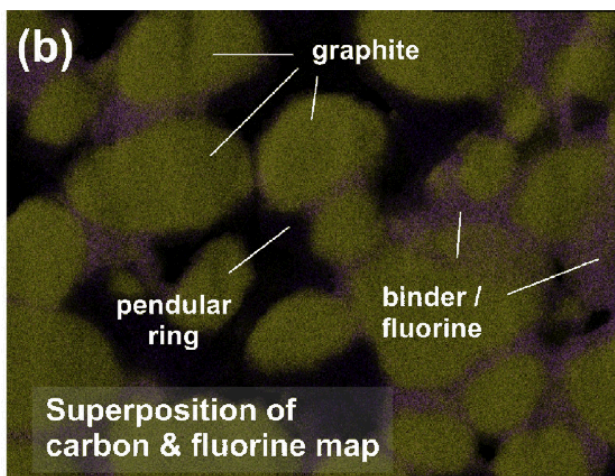
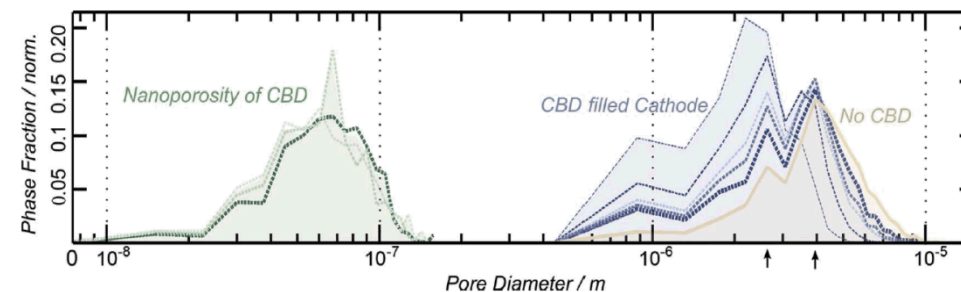
Resolving conductive binder in 3D imaging difficult

- Binder often neglected, assuming non-active void space is electrolyte
- Limited imaging results can hint at binder location

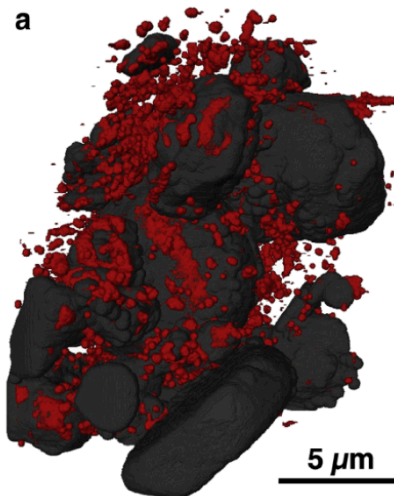
Amorphous binder is significantly nanoporous

- 47% Zielke (2015); 45% Grillet (2016)
- 5% ionic conductivity of pure electrolyte

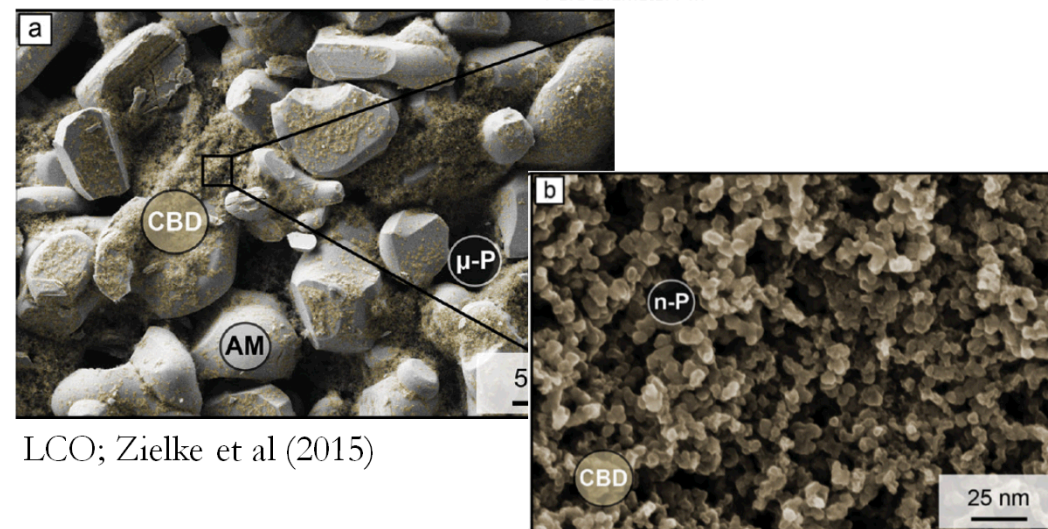
Binder weight fraction	Dense volume binder:particle	Porous volume binder:particle
0.04	0.10	0.15
0.06	0.16	0.23
0.08	0.22	0.31
0.10	0.28	0.40



Graphite; Jaiser et al. (2017)



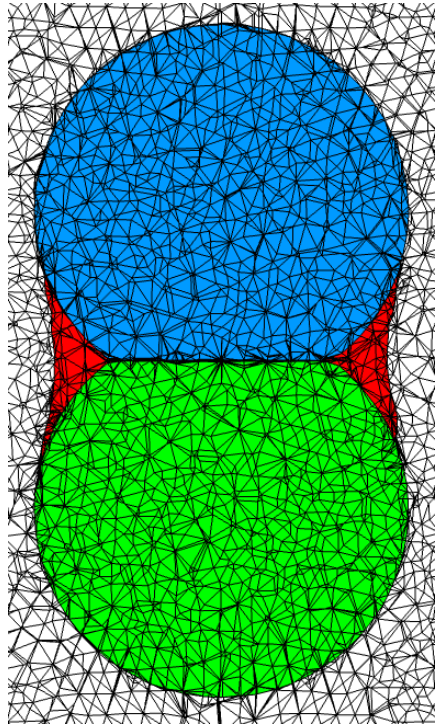
LCO; Komini Babu et al (2015)



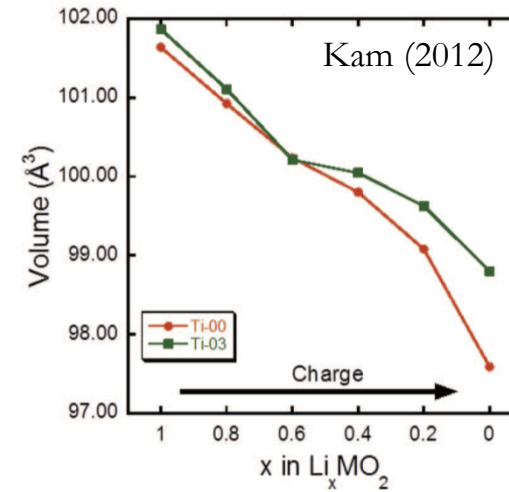
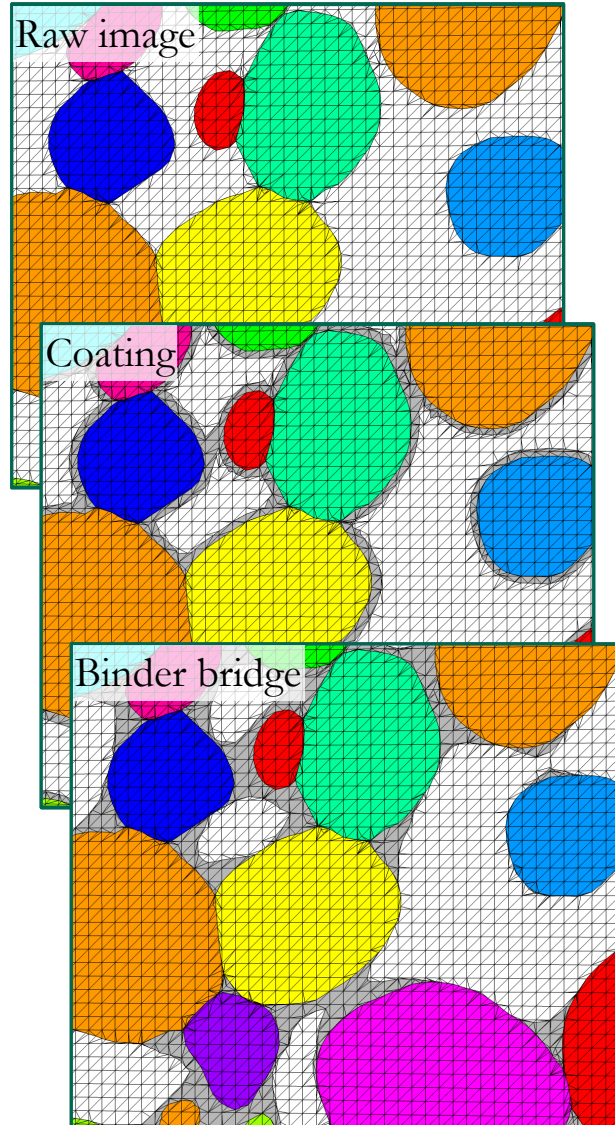
LCO; Zielke et al (2015)

How are electrode-scale properties affected by the inclusion of binder? How does the morphology matter?

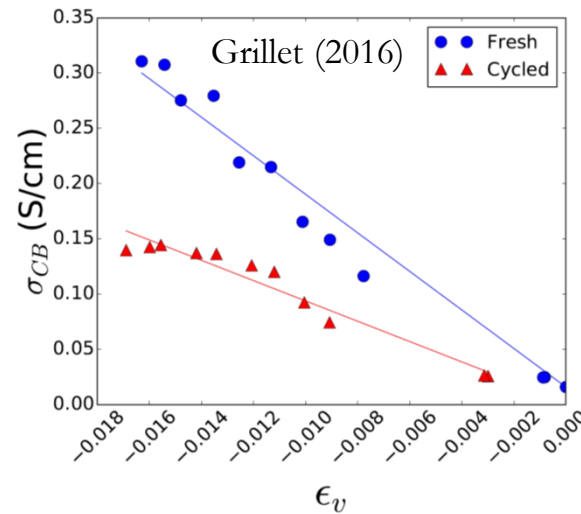
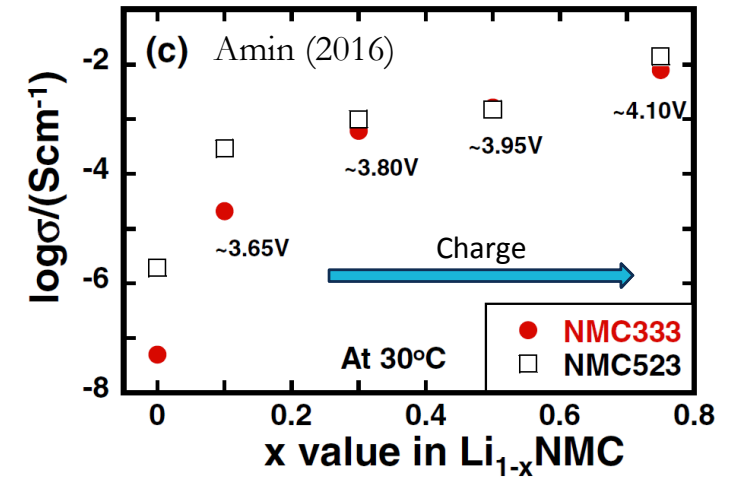
Binder bridge morphology approach



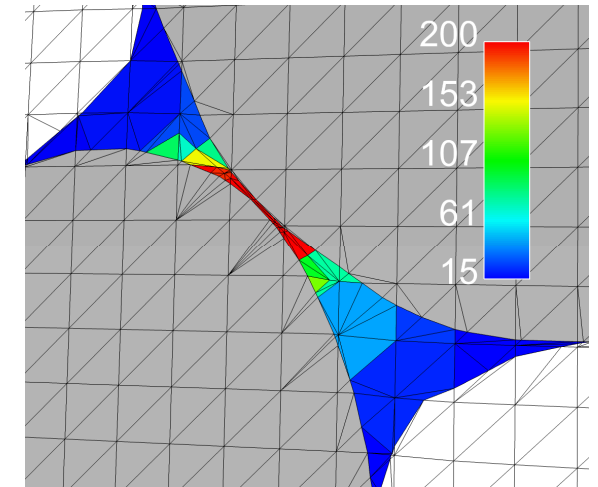
Mathematical description of "binder bridges"



NMC has lithiation-dependent properties

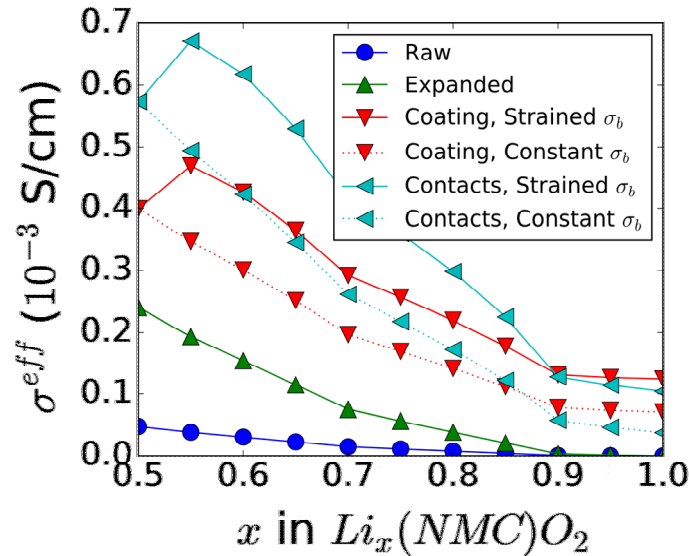


Binder has strain- (e.g. lithiation-) dependent properties



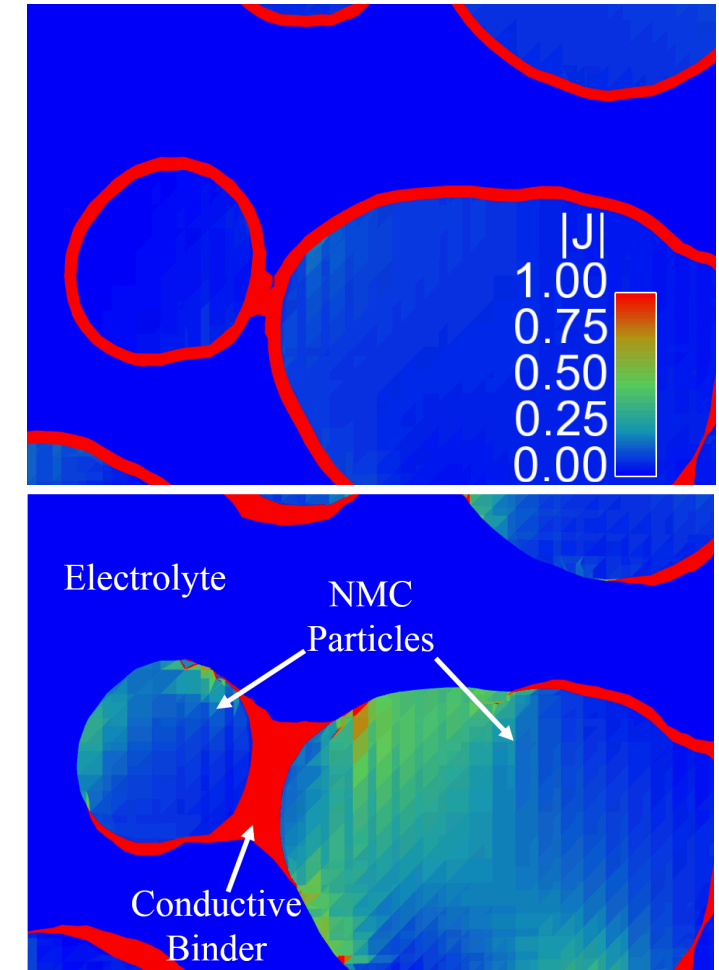
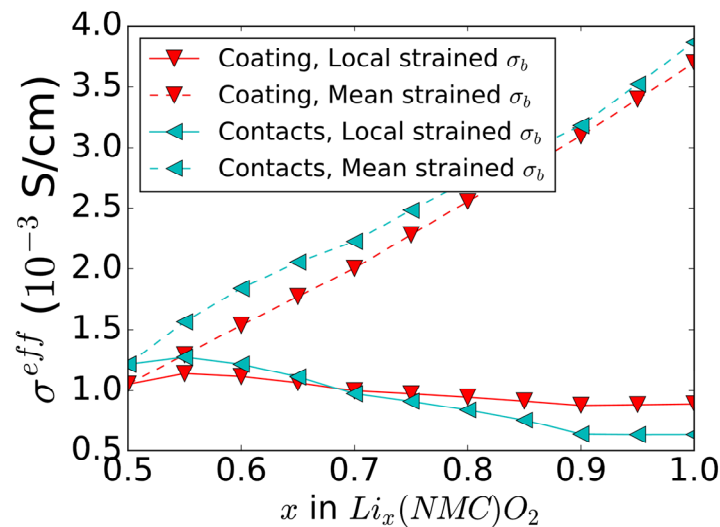
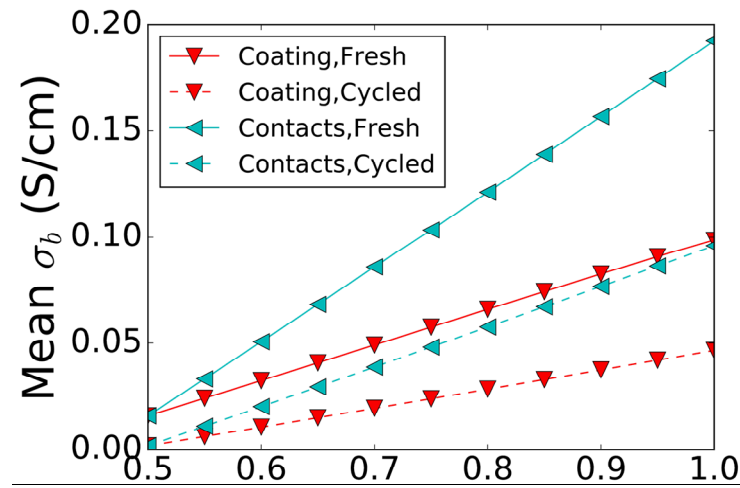
Binder bridge mimics experimental observations; properties are lithiation-dependent

Effect of including binder on effective properties



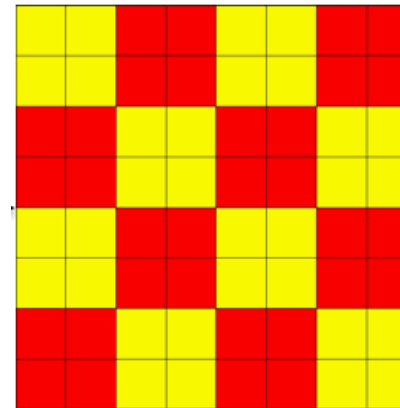
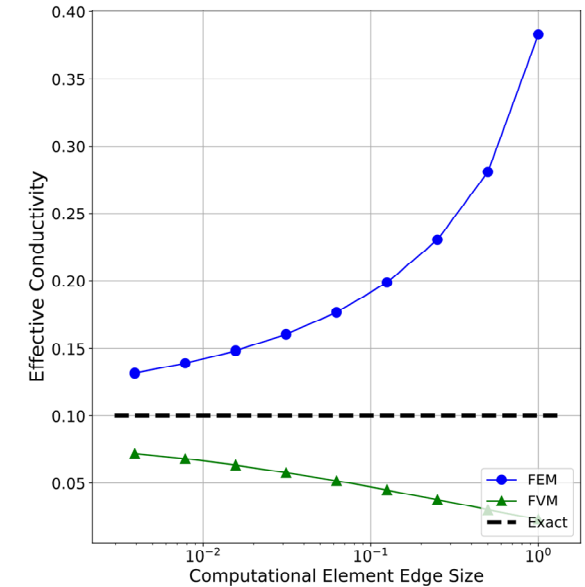
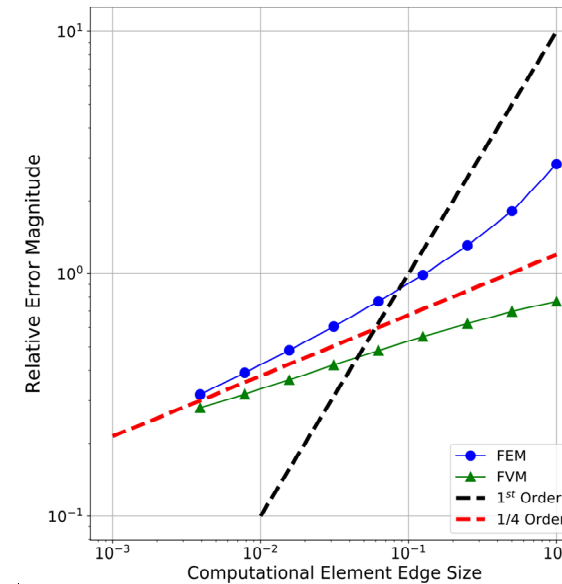
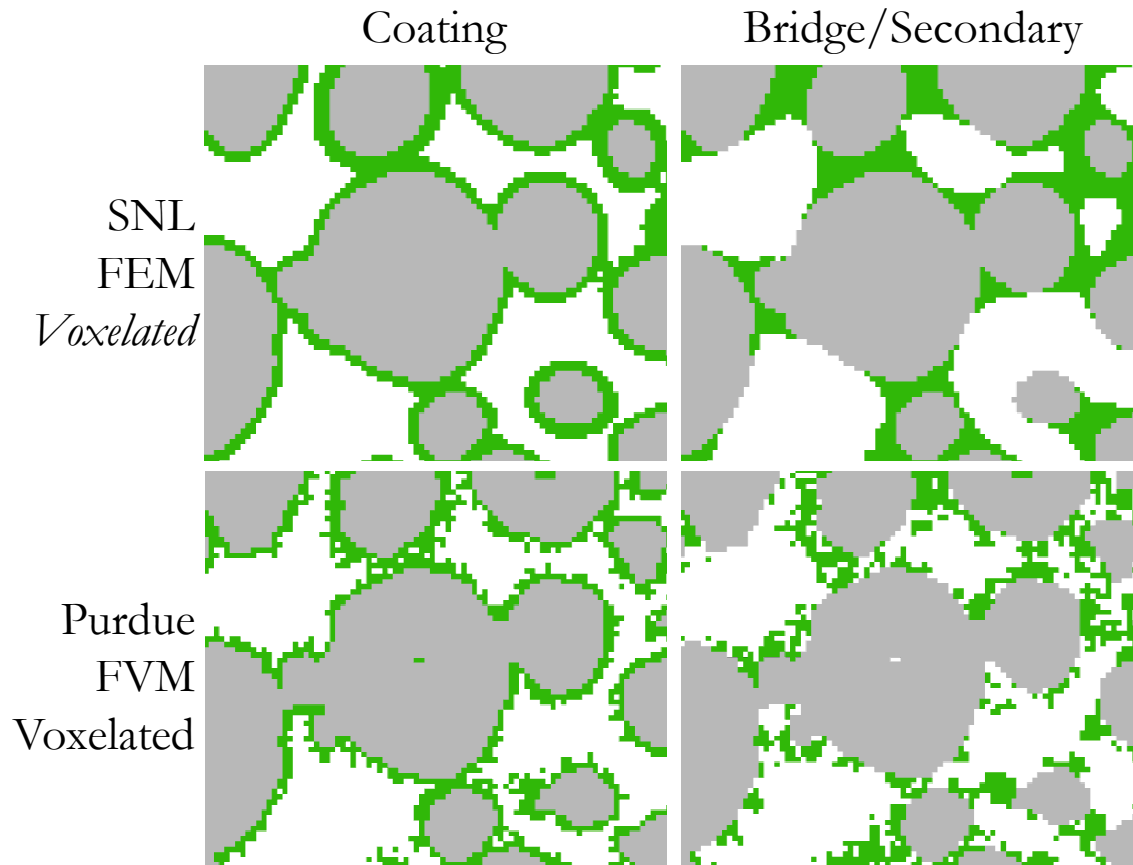
Presence of binder increases electrical conductivity. Mechanics further increase conductivity.

Binder morphology and current localization matters. Binder bridge forces current flow through particles.



Binder morphology and mechanical coupling have a significant impact on effective properties; localization matters!

What about other morphologies and numerical methods

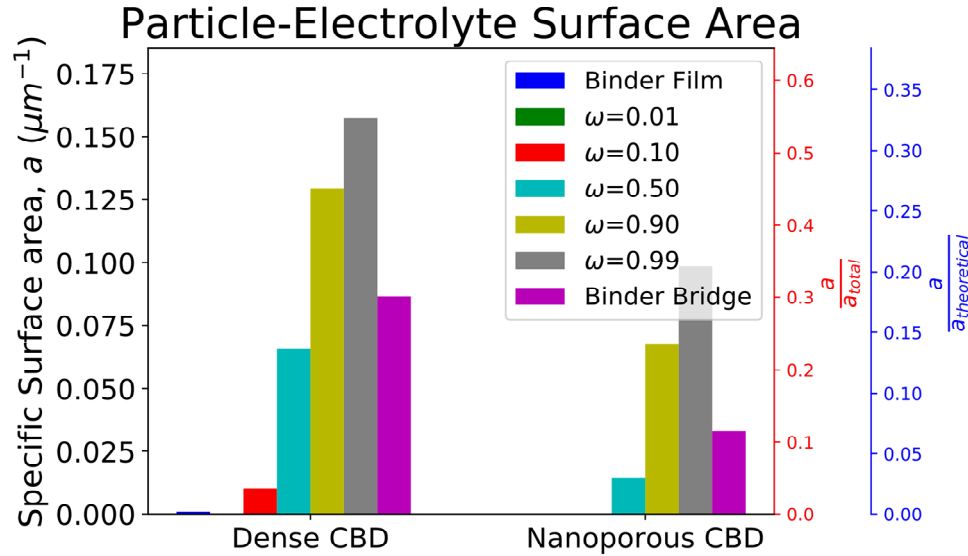


Both methods converge on voxelated meshes, but:

- At slow rates (1/4-order)
- From different directions
- Unrefined results 10x different

Care must be taken when comparing results generated using different numerical methods; likely not converged!

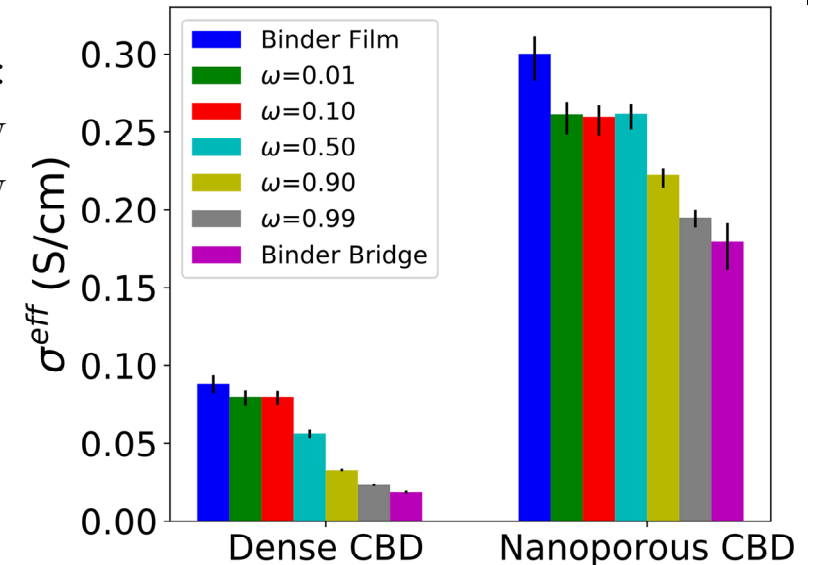
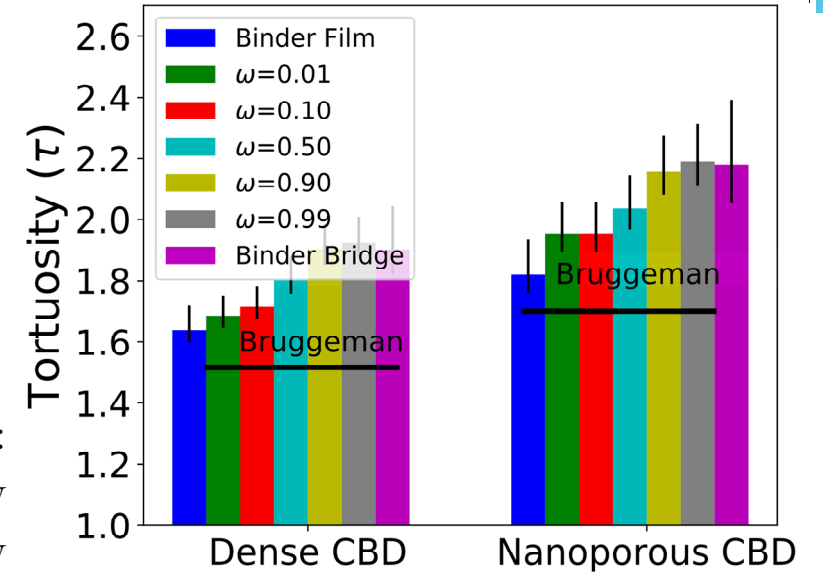
Porous binder and morphology considerations



- More particle surface area available with non-uniform morphologies
- Nanoporous binder decreases bare particle surface area, but binder area is porous
- Surface area much less than theoretical

- Non-uniform binder:
- Increases tortuosity
 - Decreases conductivity

- Nanoporosity:
- Increases tortuosity
 - Increases conductivity

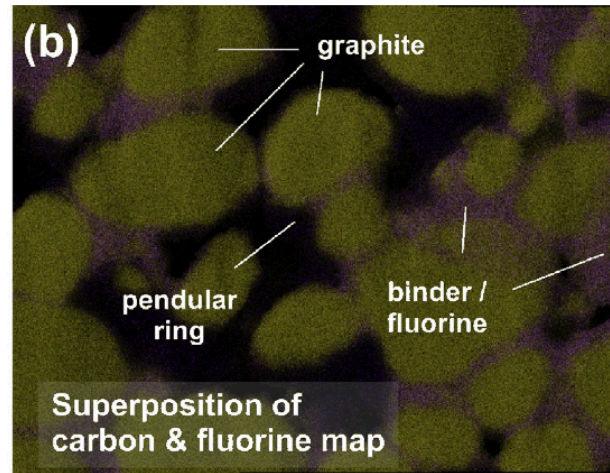


Limiting cases of both morphology methods show similar (but not identical) behavior; nanoporosity is important!

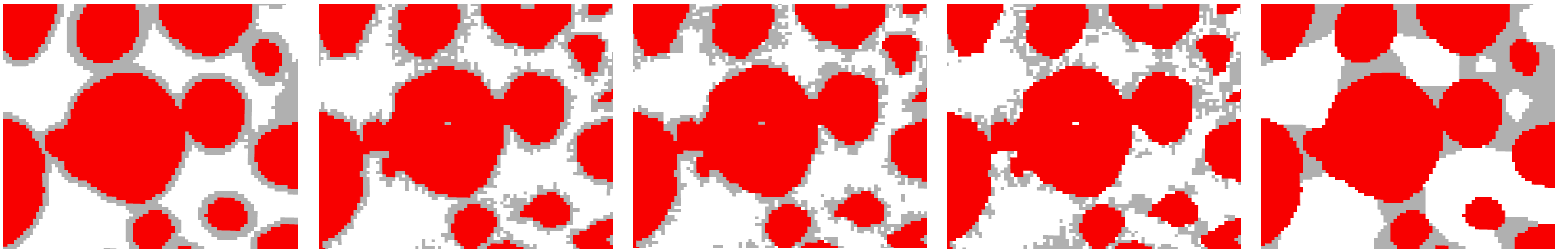
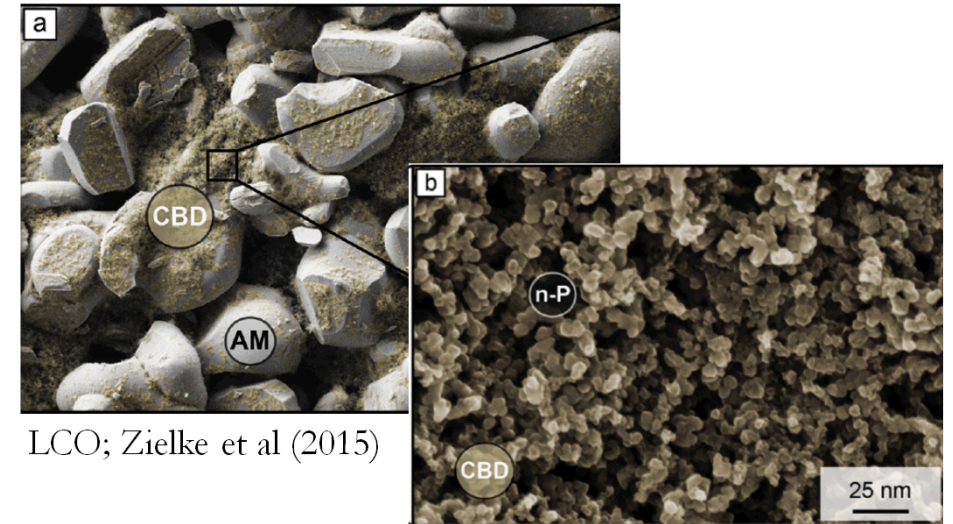
Challenges with using CT mesoscale data

3D CT image data:

- Expensive
- Time consuming
- Conductive binder not visible



Graphite; Jaiser et al. (2017)



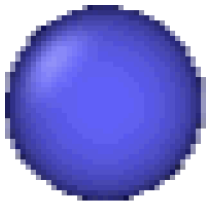
Hypothesis: Use DEM simulations to create AM+CBD mesostructures and CDFEM for physics predictions

Discrete Element Method (DEM) setup

LAMMPS solves Langevin equation for inertial Brownian motion

- 94:3:3 AM:CB:PVDF weight ratio
- $(\sim 100 \mu\text{m})^3$ domain at 50% porosity
- 1,000 AM particles, 800,000 CBD particles

Active Matter (AM): NMC particle [10 microns] Carbon black + PVDF binder (CBD) particle [500 nm]



Inter-particle interactions

$$F_{\text{total}} = F_{\text{JKR}} + F_{\text{visc}} + F_{\text{brown}}$$

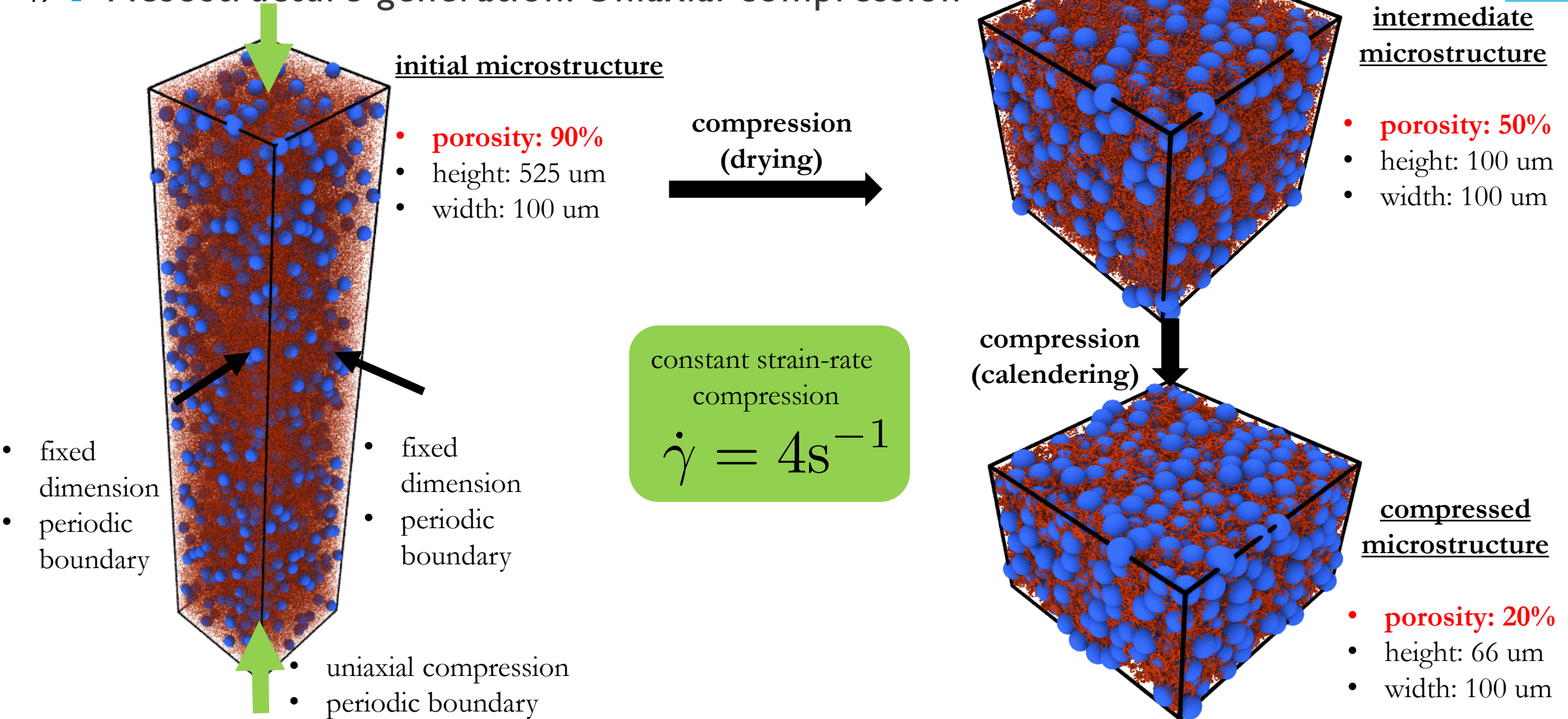
$$F_{\text{JKR}} = \left(\frac{4E\delta^3}{3a} - 2\pi\delta^2 \sqrt{\frac{4\gamma_c E}{\pi\delta}} \right)$$

$$F_{\text{visc}} = 6\pi\eta av$$

$$F_{\text{brown}} = \sqrt{2mk_B T \eta \delta(t)}$$

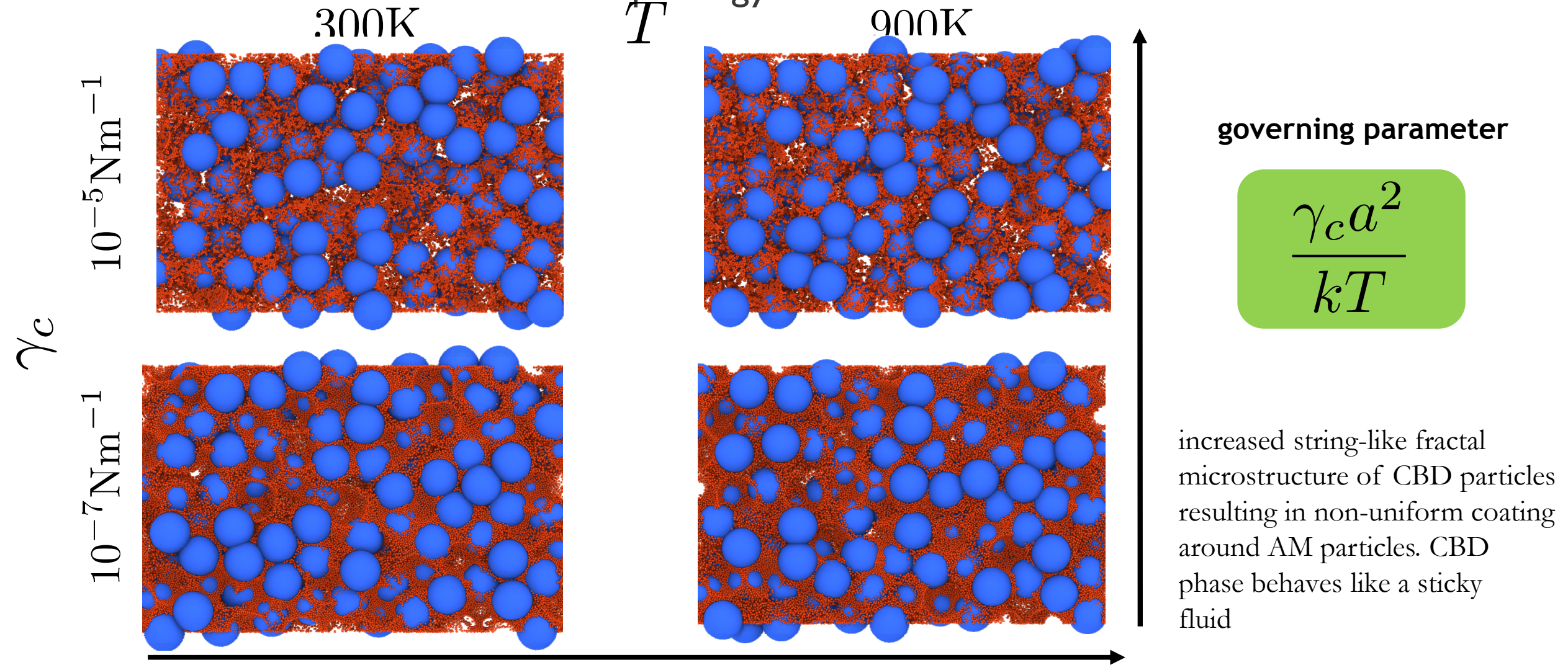
Granular + Brownian motion to enable study of AM consolidation and CBD aggregation

Mesostructure generation: Uniaxial compression



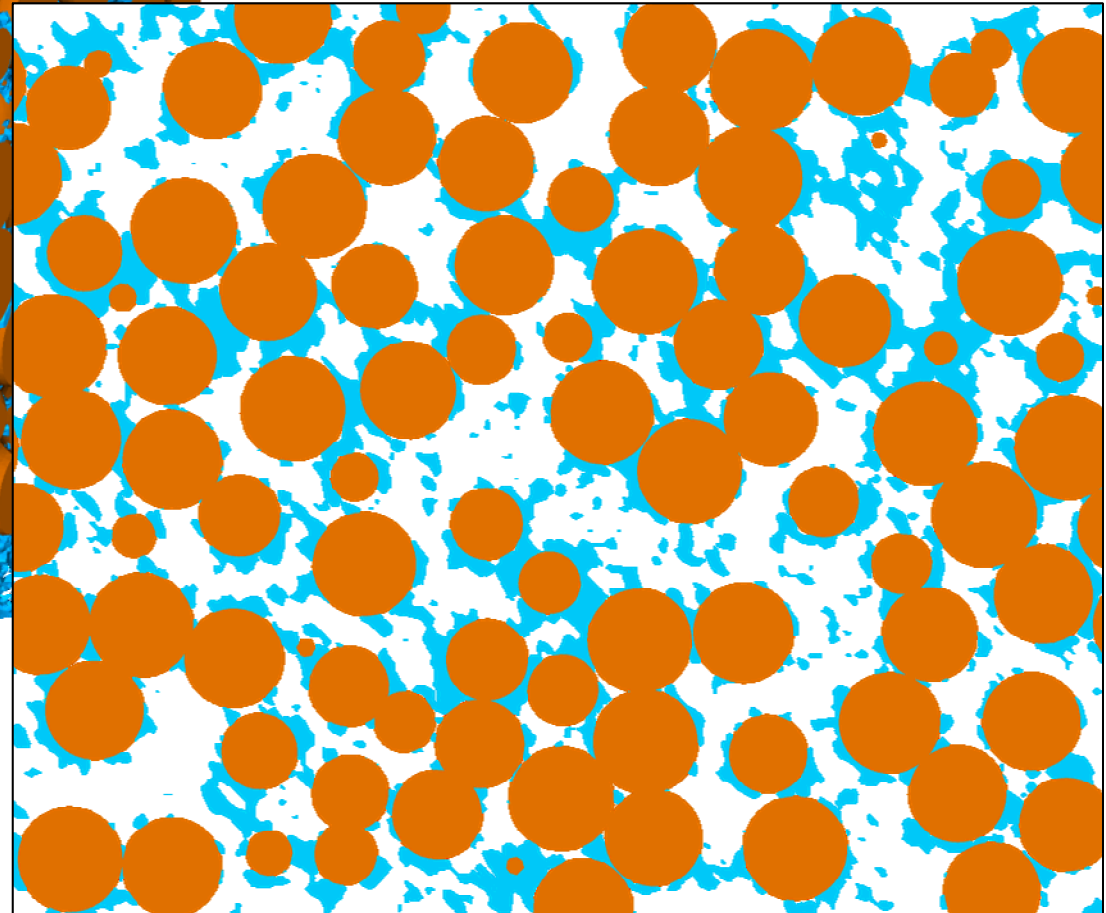
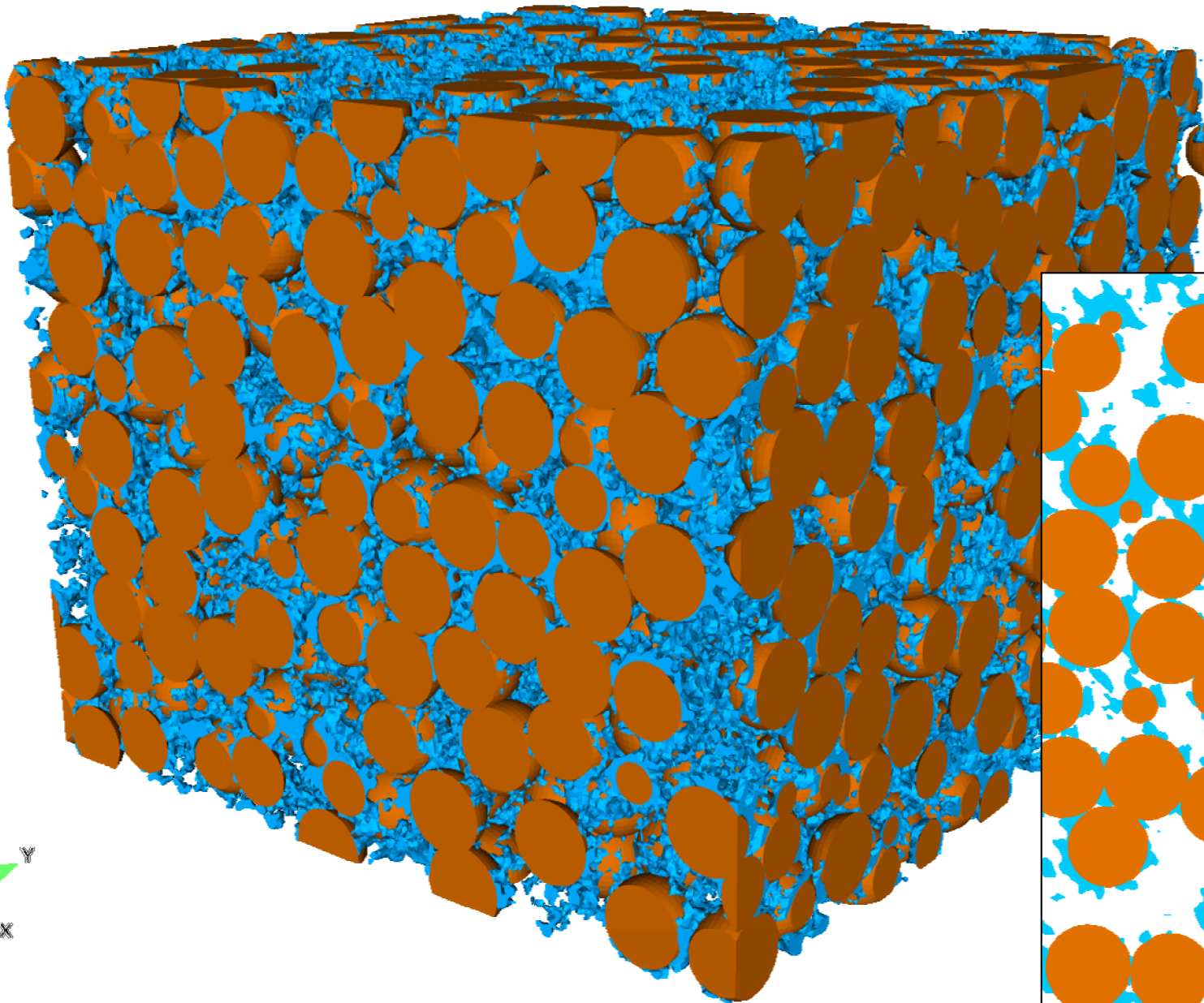
Uniaxial compression represents both drying and calendering

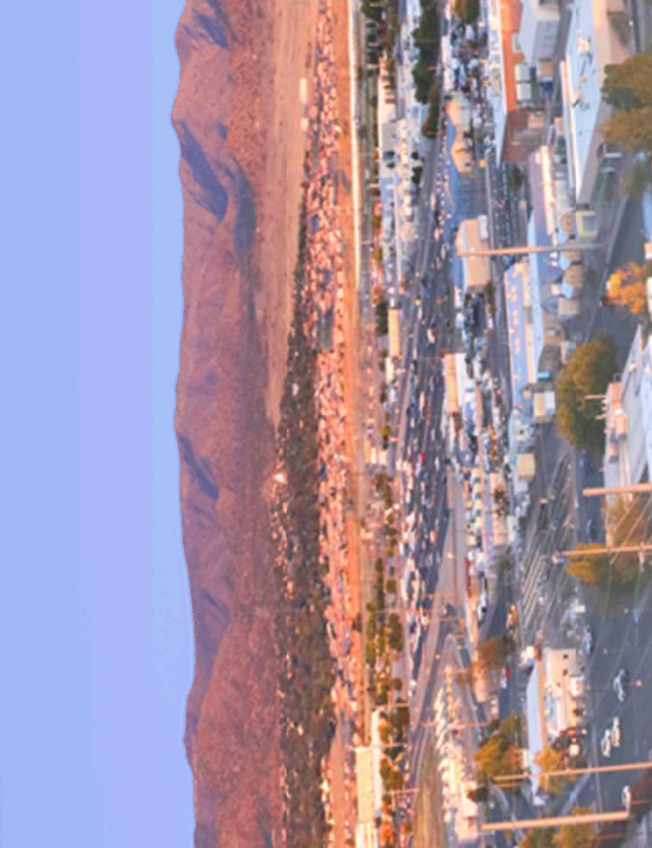
Role of cohesion in CBD morphology



increased CBD diffusion results in a more uniform coating around AM particles

Cohesive surface energy drastically alters CBD morphology





NMC cathode effective property prediction and upscaling



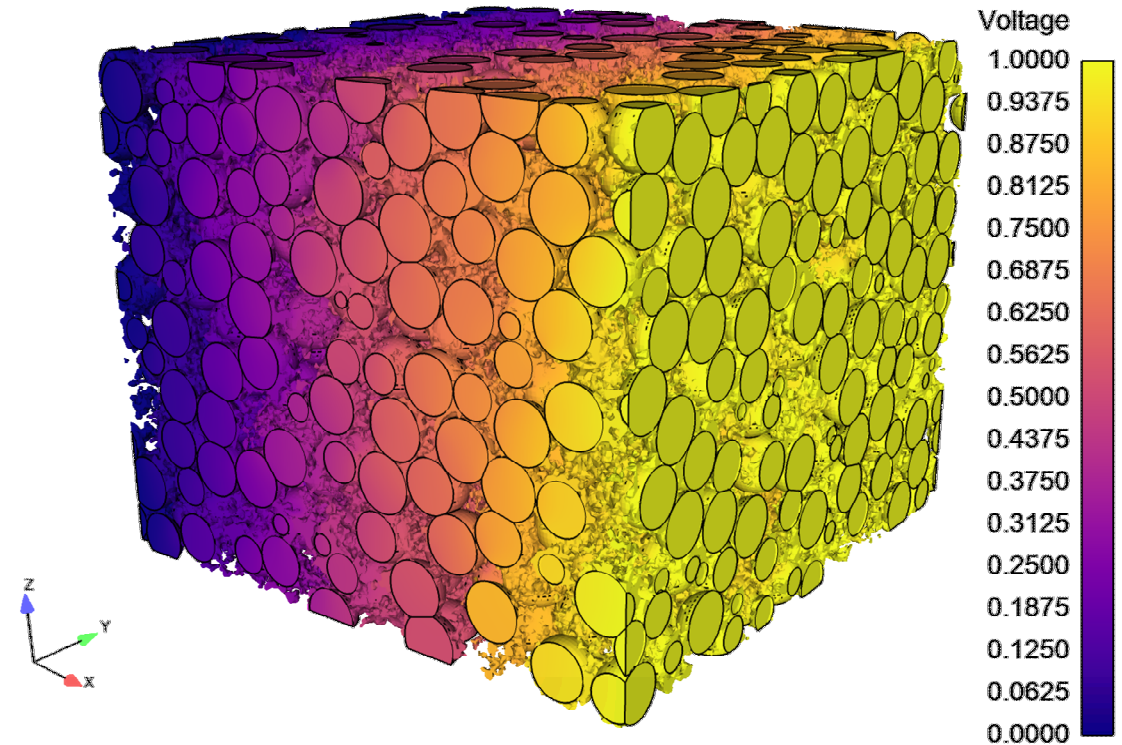
Effective electrode property calculations

Calculate effective transport properties for upscaling

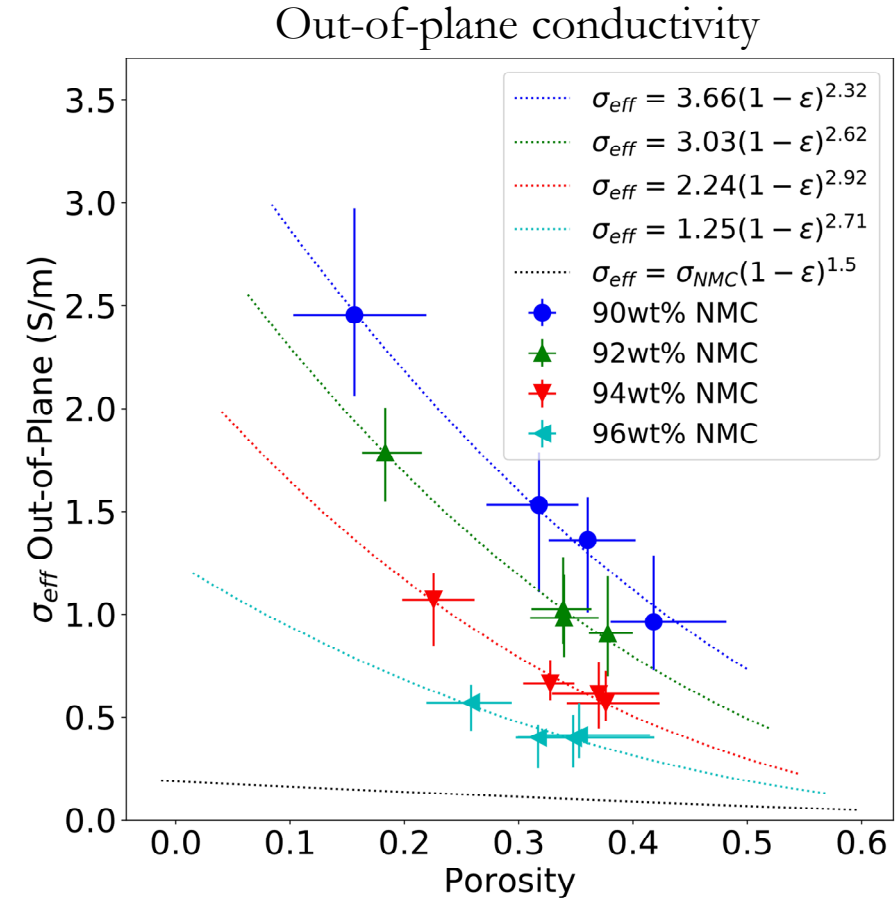
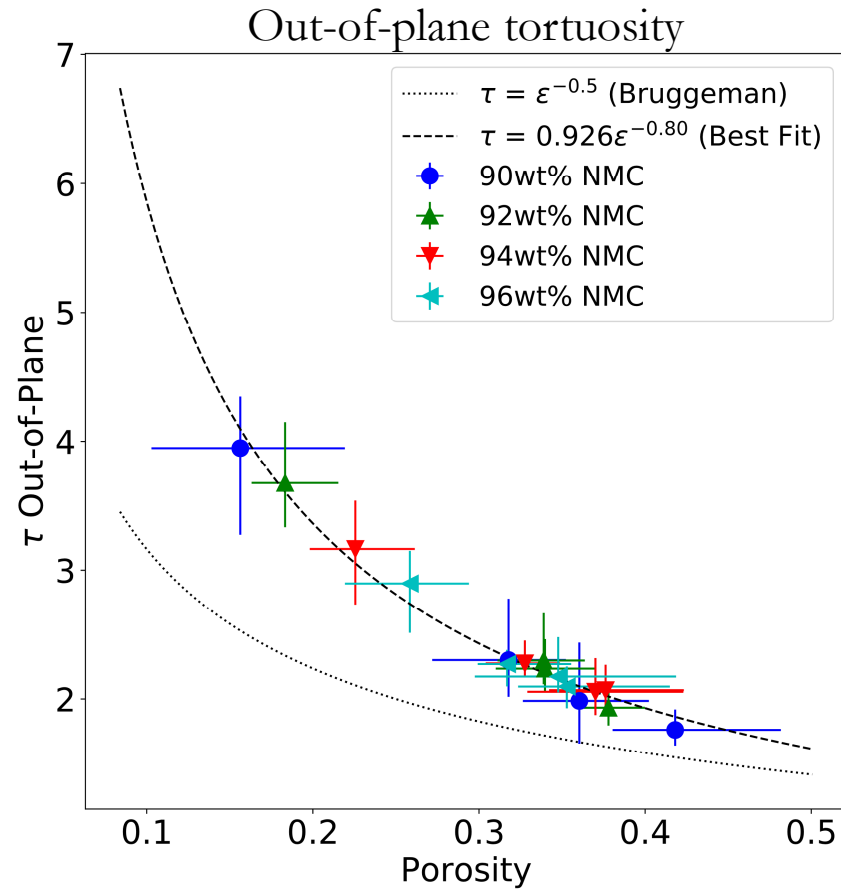
- Particle specific surface area
- Electrical conductivity
- Tortuosity

NMC image data from Ebner (2013)

- 90, 92, 94, 96 wt% NMC (remainder 1:1 CB:PVDF)
- 0, 300, 600 & 2000 bar calendering
- $100\text{ }\mu\text{m} \times 100\text{ }\mu\text{m} \times 60\text{ }\mu\text{m}$ domain (20 realizations eac
- Binder bridge (porous) morphology approach

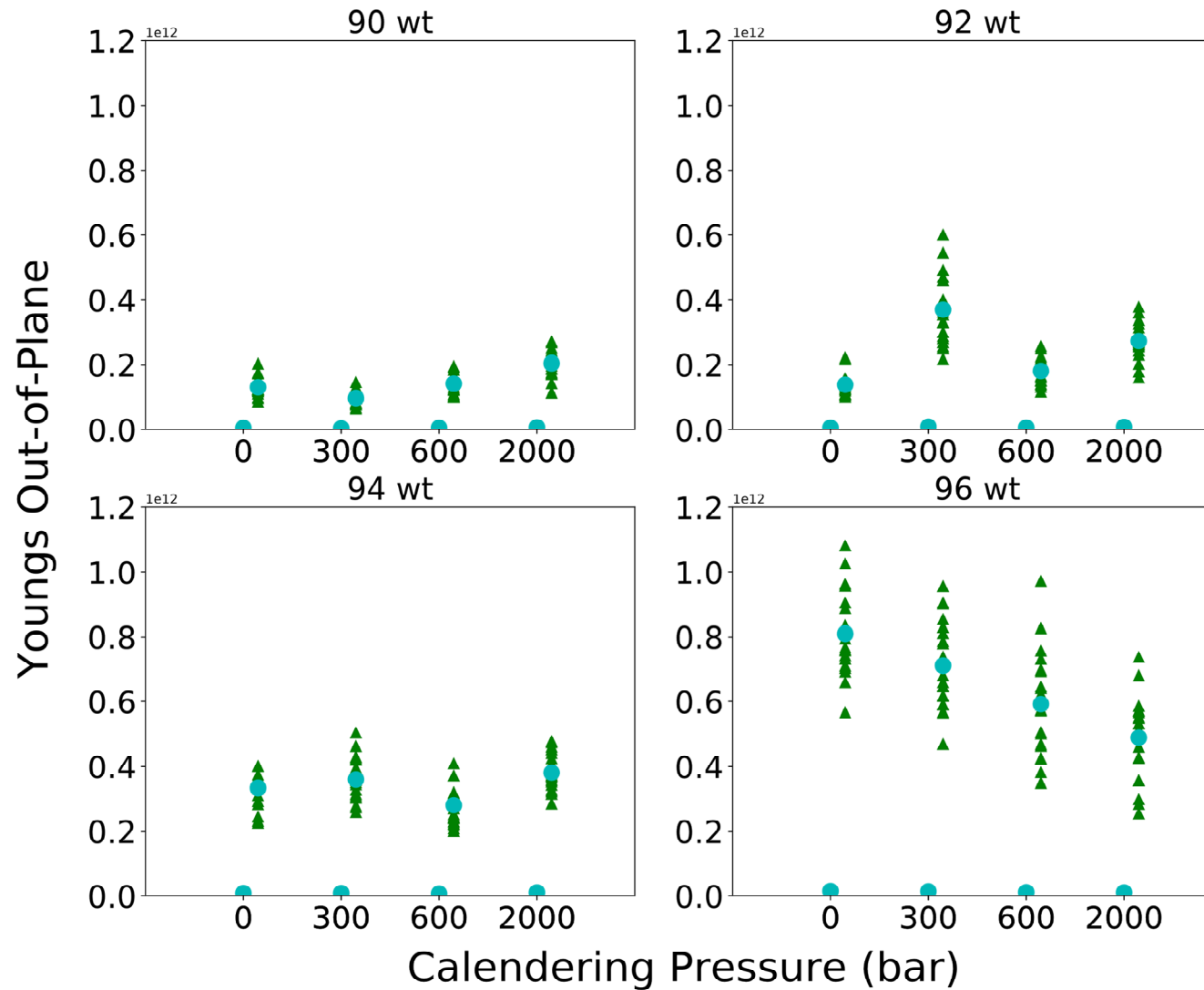


Effective properties are an important first step for upscaling mesoscale data

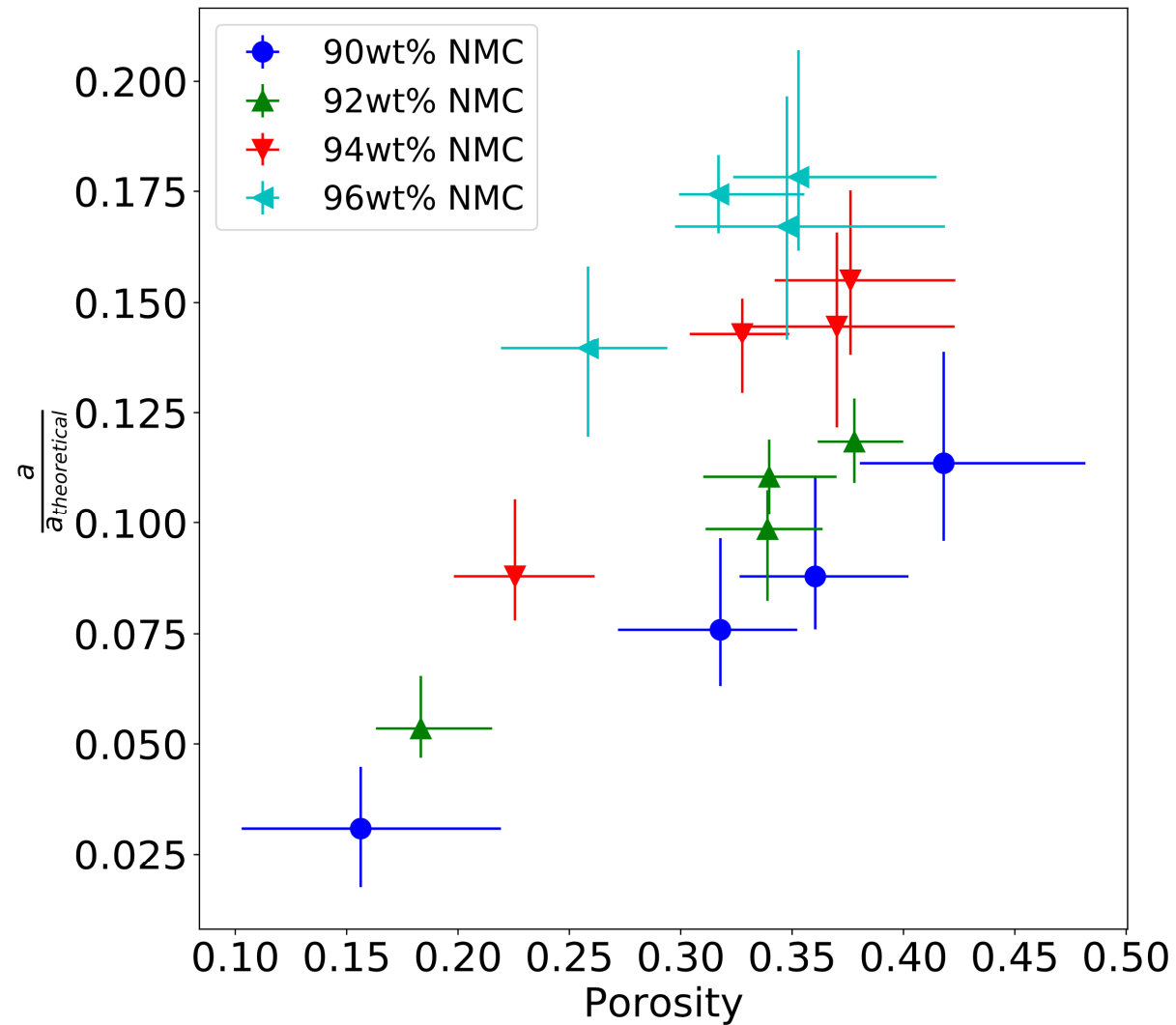


Bruggeman relationships must be re-calibrated to fit simulated data

Effective electrode property calculation results – Mechanical



Stiffness increases with NMC content but also varies with calendaring

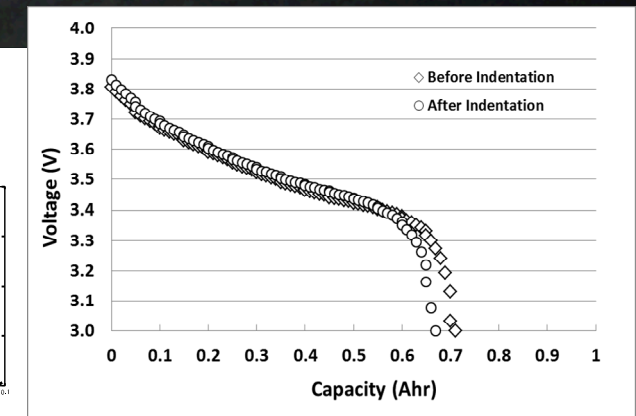
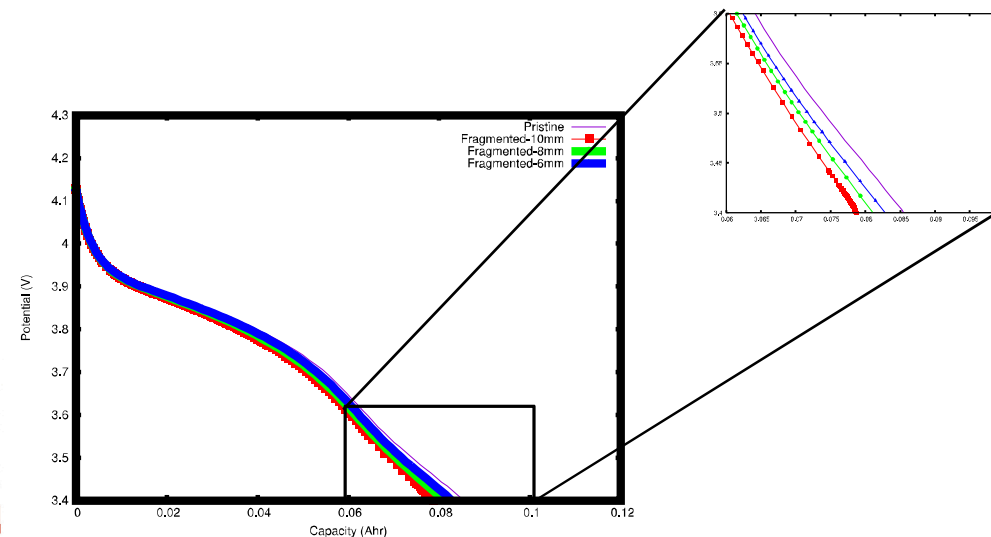
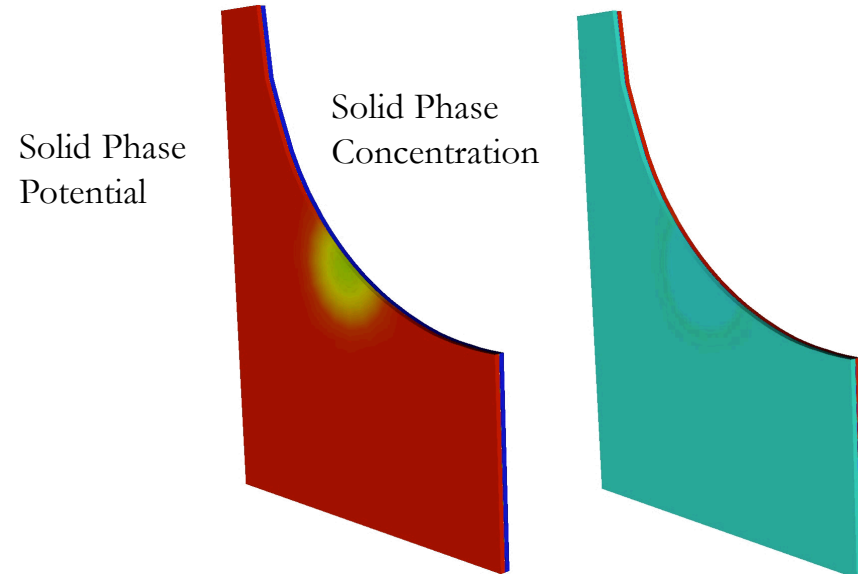
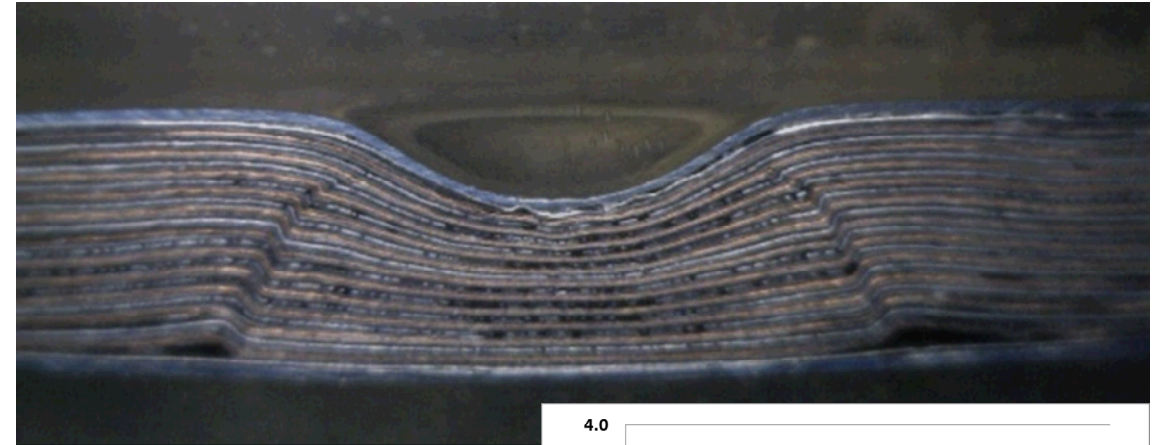


Binder morphology significantly reduces particle surface area available for chemical reaction

Upscaling effective properties for pouch-level modeling

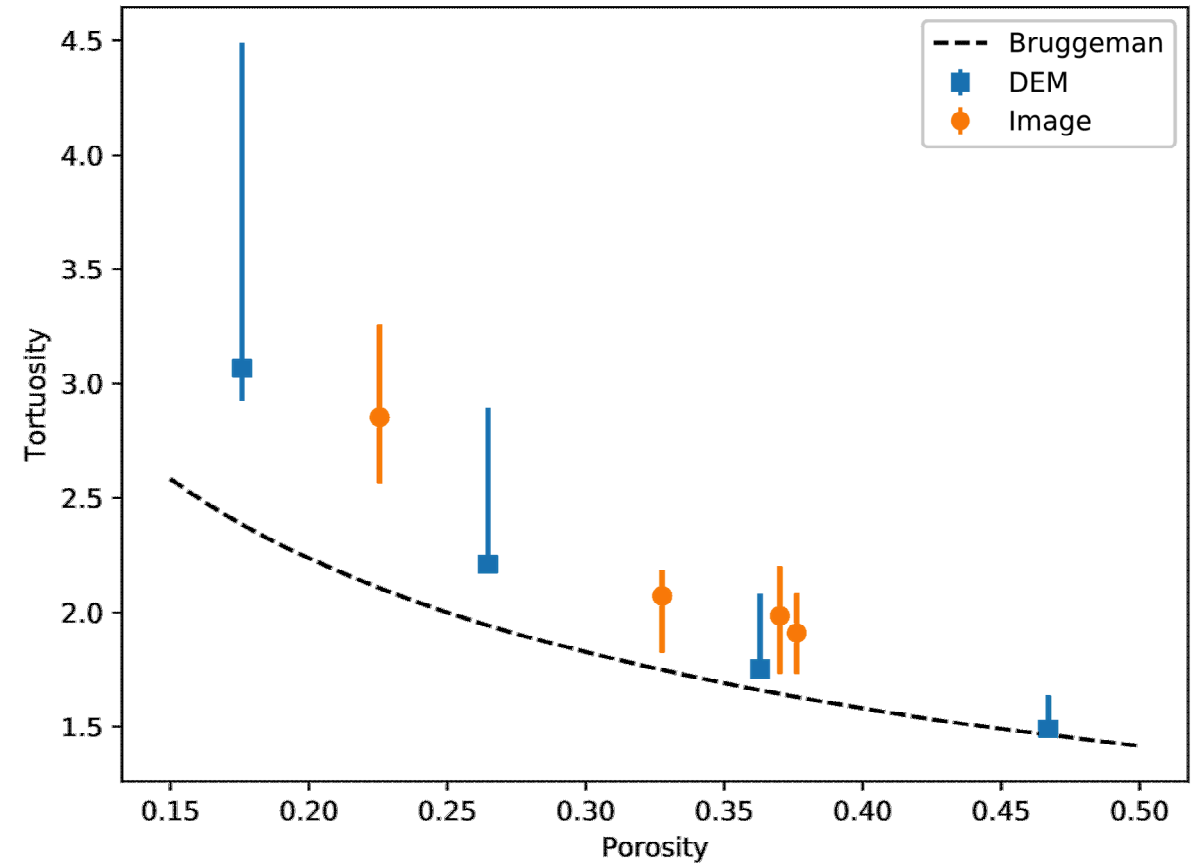
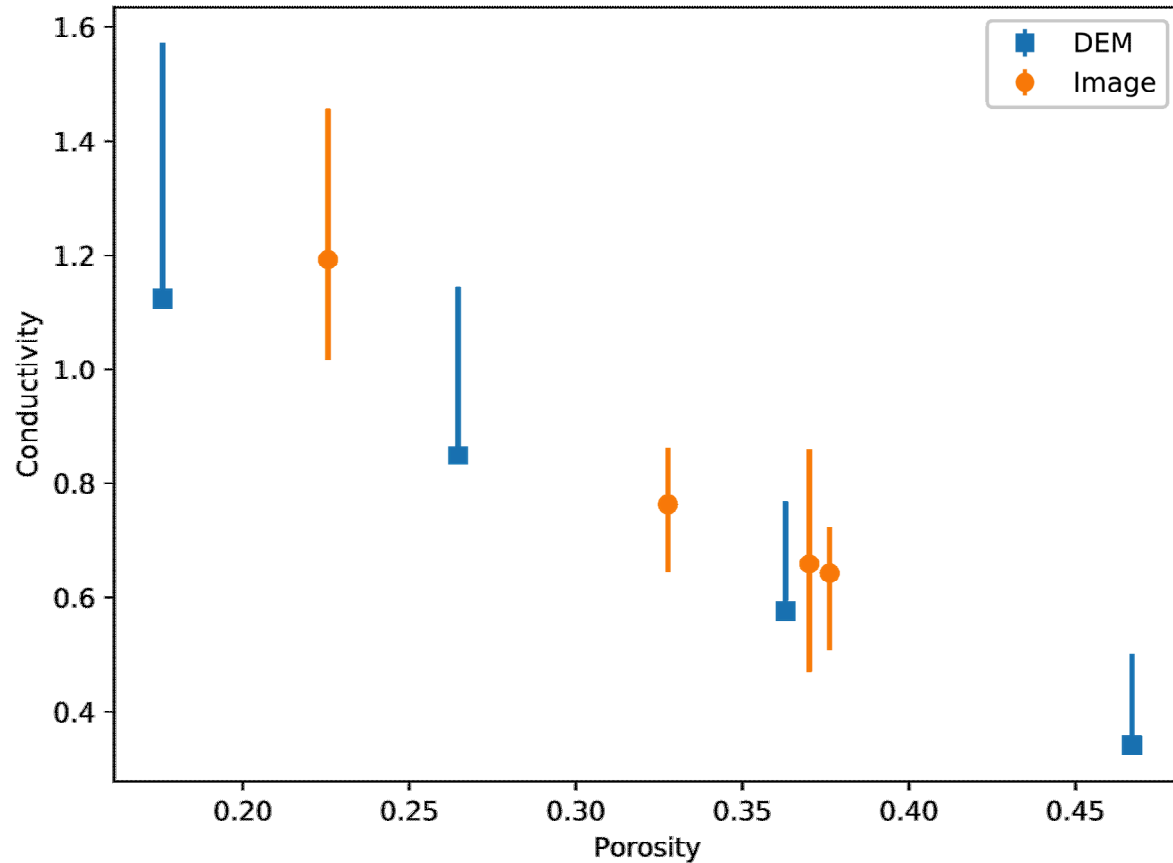
Functional forms as function of porosity integrated into VIBE/OAS and compared to simulation

- 10 cell pouch with 80 μm electrodes
- Indented with 13 mm diameter spherical indenter
- Modeled mechanics of indentation (porosity change)
- Simulated 1C discharge



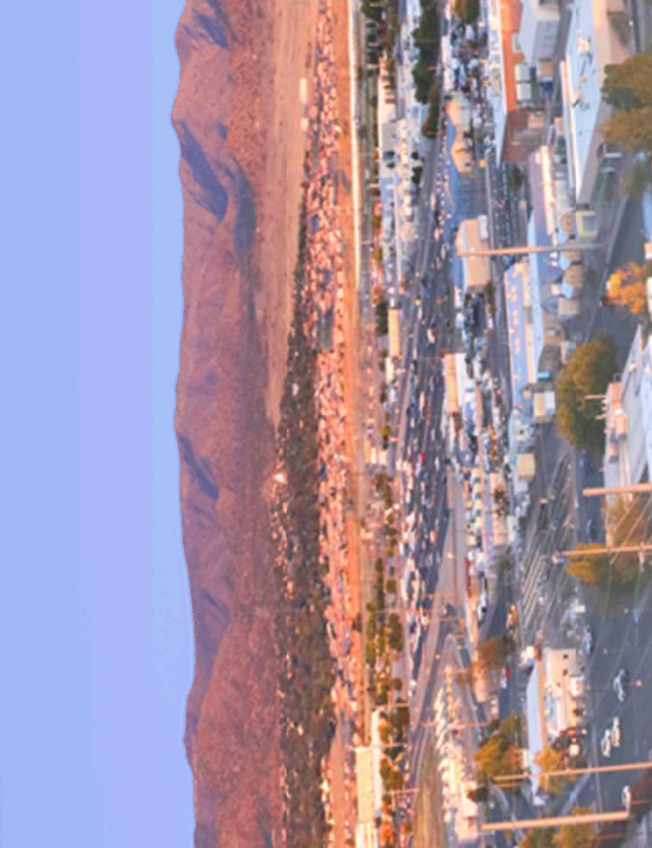
Pouch simulation with effective properties shows similar-scale capacity decrease; demonstrates mesoscale integration

Comparison of image- and DEM-based mesostructures



All simulations for $T=600$, $\gamma=10^{-5}$, AM=94 wt%, 0 bar calendaring

Calendaring $\rightarrow$ lower porosity $\rightarrow$ more CBD connectivity $\rightarrow$ higher conductivity and tortuosity



Electrochemical-mechanical discharge simulations of NMC half-cells



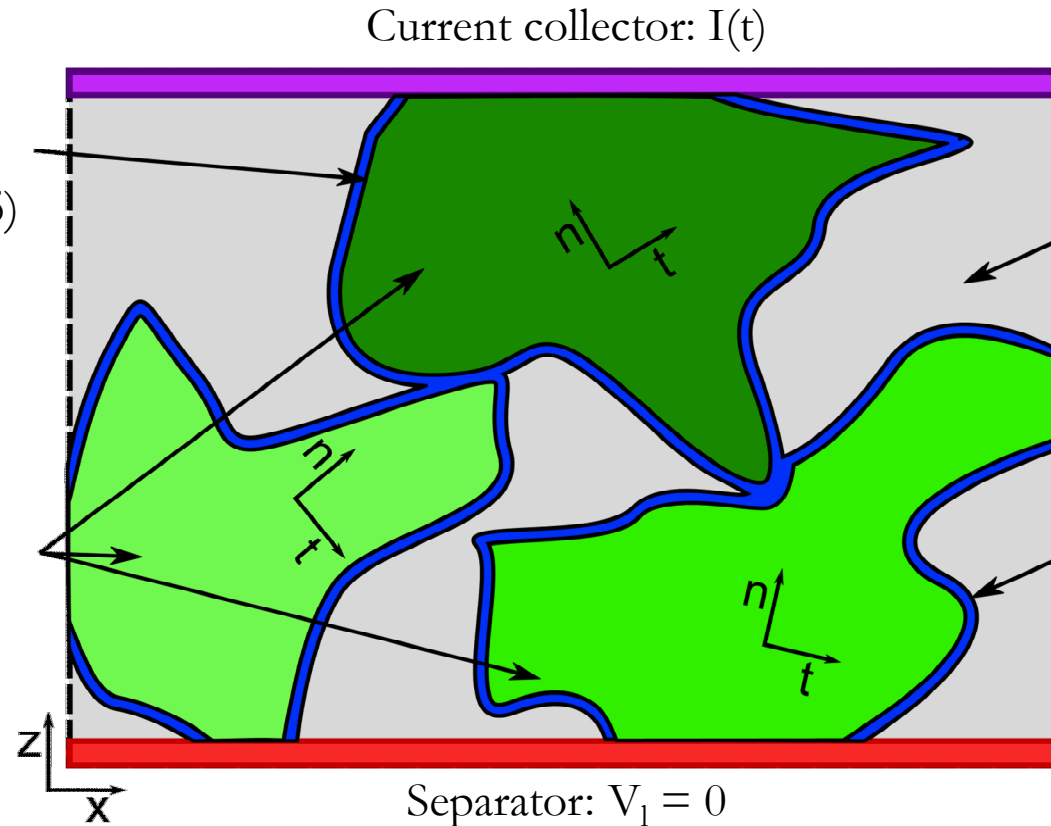
Coupled electrochemical-mechanical half-cell discharge simulations

Particle Interface:

- Butler-Volmer reaction
- OCV from Smekens (2015)

Particles:

- Species – Li tr
 - Chemical
 - Stress pc
- Electrical – ϕ
- Mechanics - E
 - Li-induc



Electrolyte:

- Species – Li^+ transport
 - Nernst-Planck fluxes
 - Electroneutrality for PF_6^-
- Current conservation

Conductive binder:

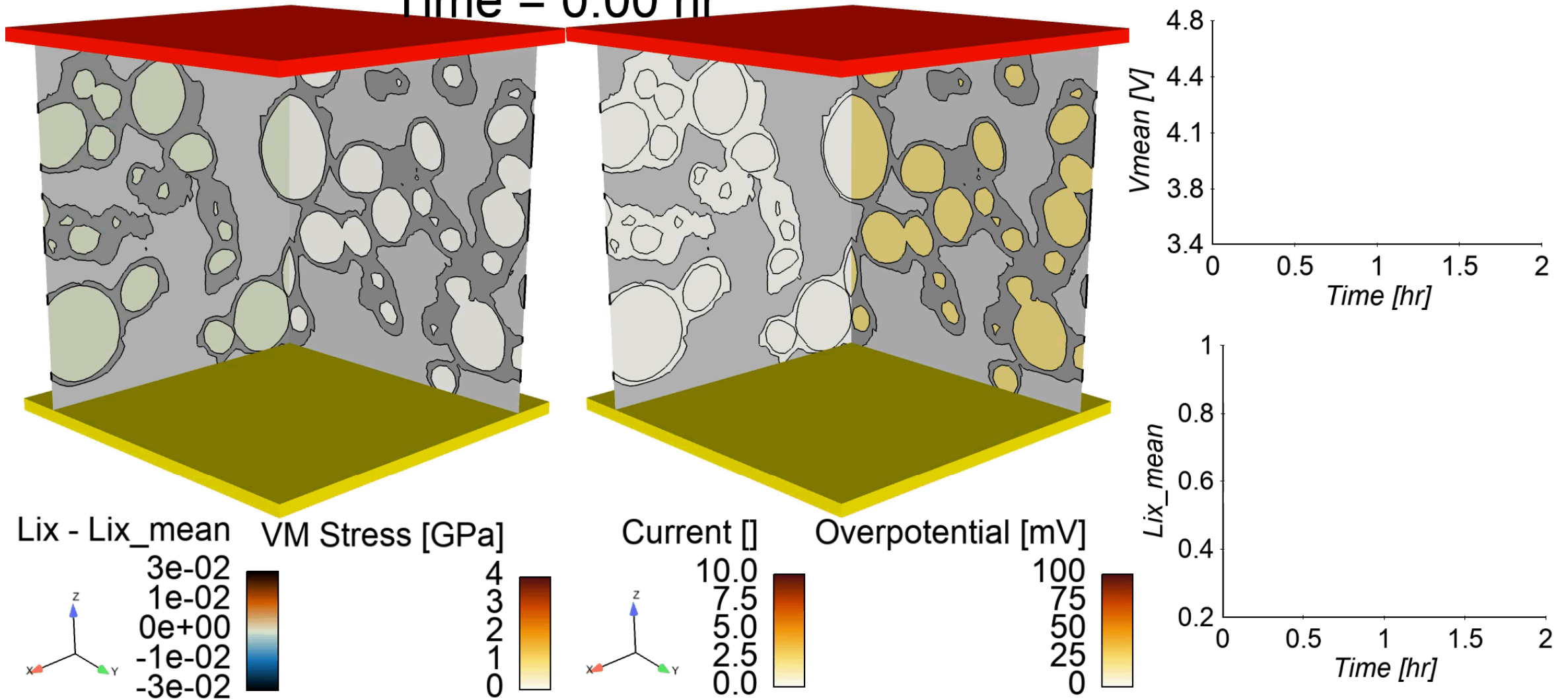
- Species – Porous Li^+ transport
- Electrical
 - Solid: Porous Ohm's law
 - Strain-dependent electrical conductivity
 - Liquid: Ionic conservation & electroneutrality
- Mechanics – Elastic

Mathematical formulation builds off of Mendoza (2016) LCO studies

Predictions of discharge curves, effects of mechanics, rate effects, and spatial variations in performance

Demonstration of NMC half-cell discharge simulation at C/2

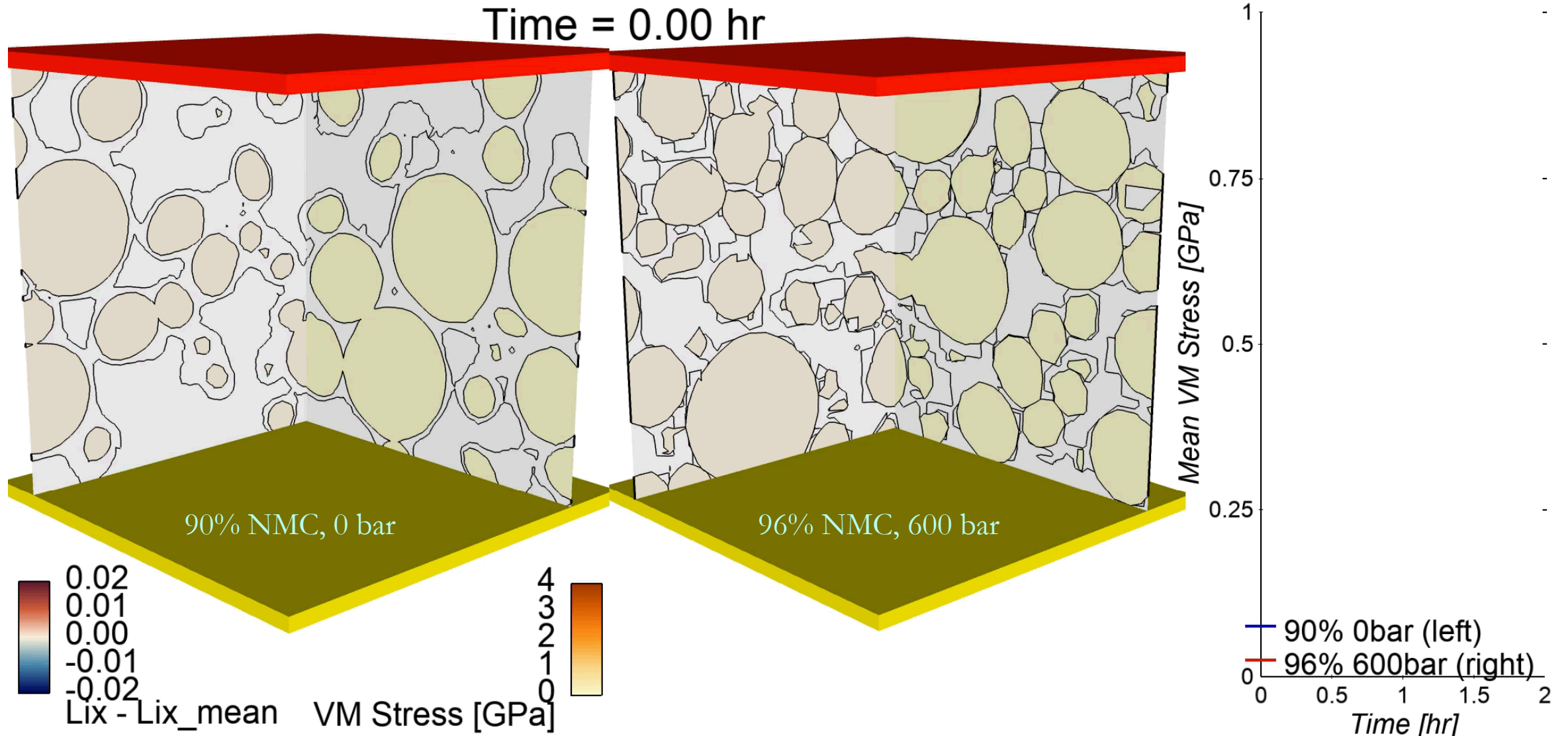
Time = 0.00 hr



Coupled electrochemical-mechanical simulation yields detailed insight, predicts electrode-scale response

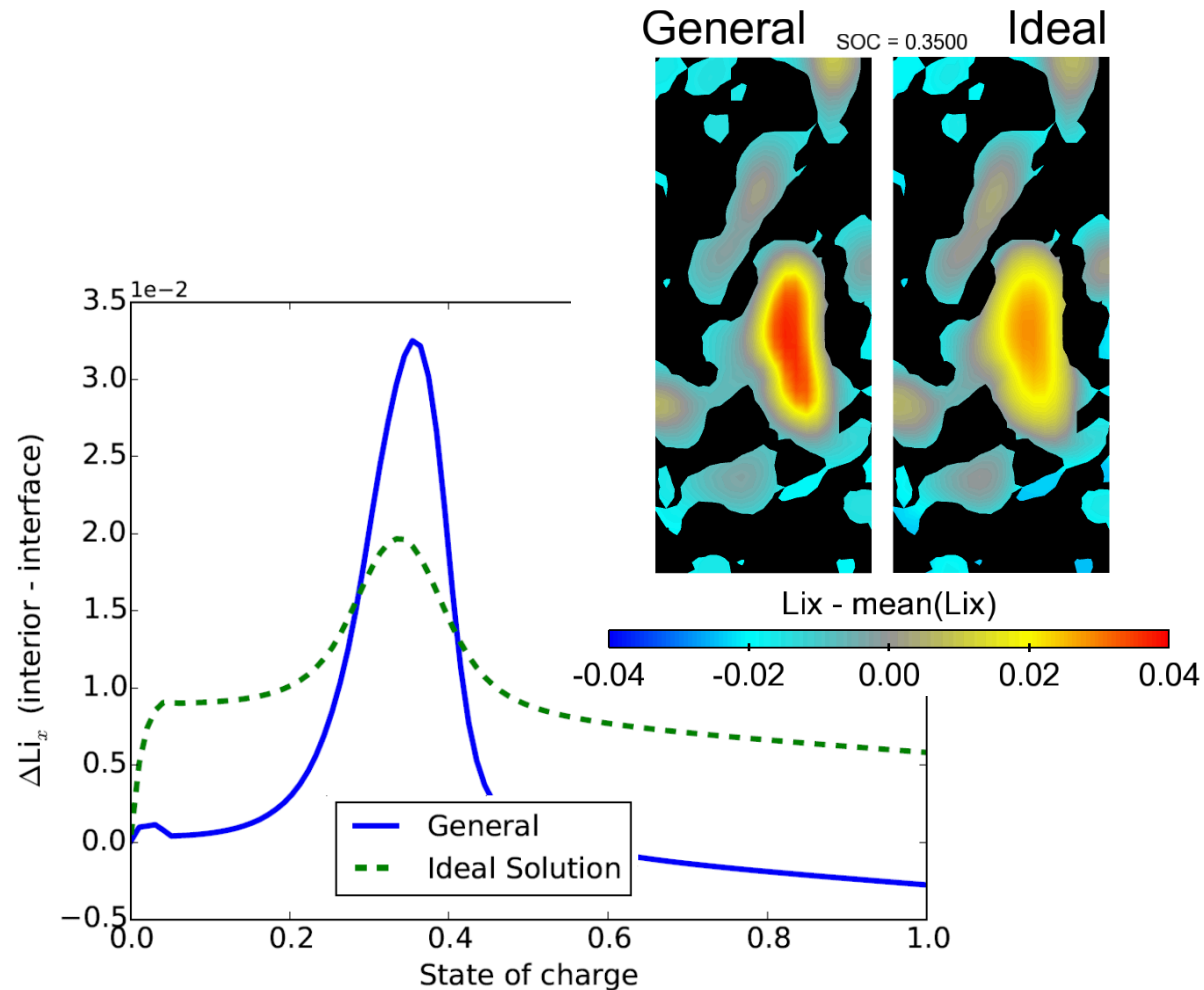
Effect of solids loading, calendering pressure

Time = 0.00 hr

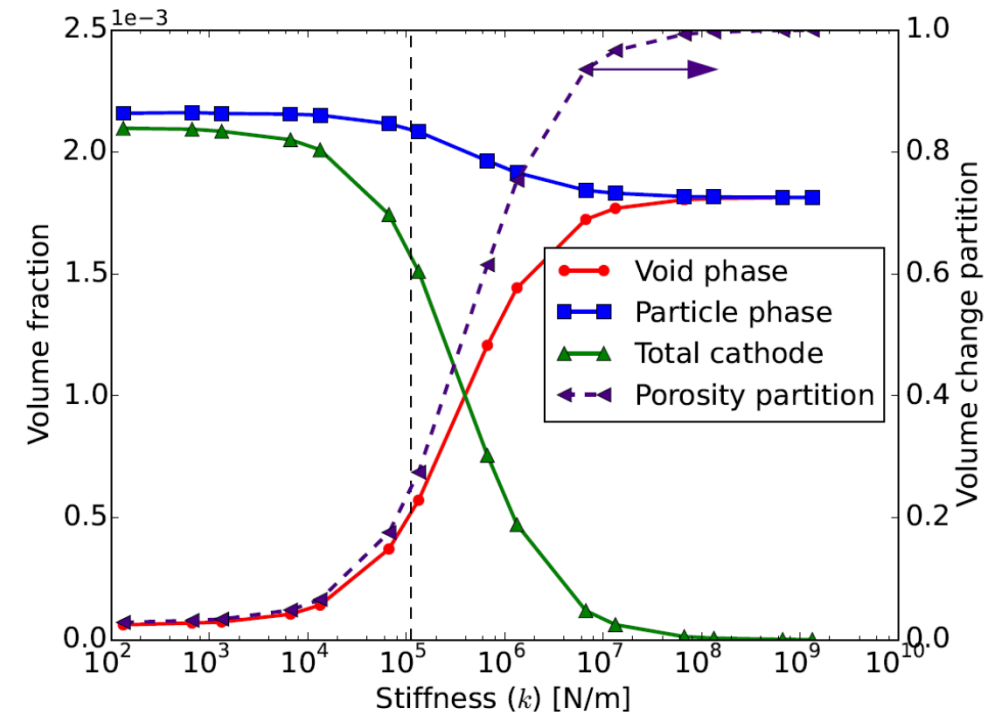


Significantly higher mechanical stresses for higher NMC loadings and higher calendering pressures

What can you learn from coupled half-cell simulations? LCO

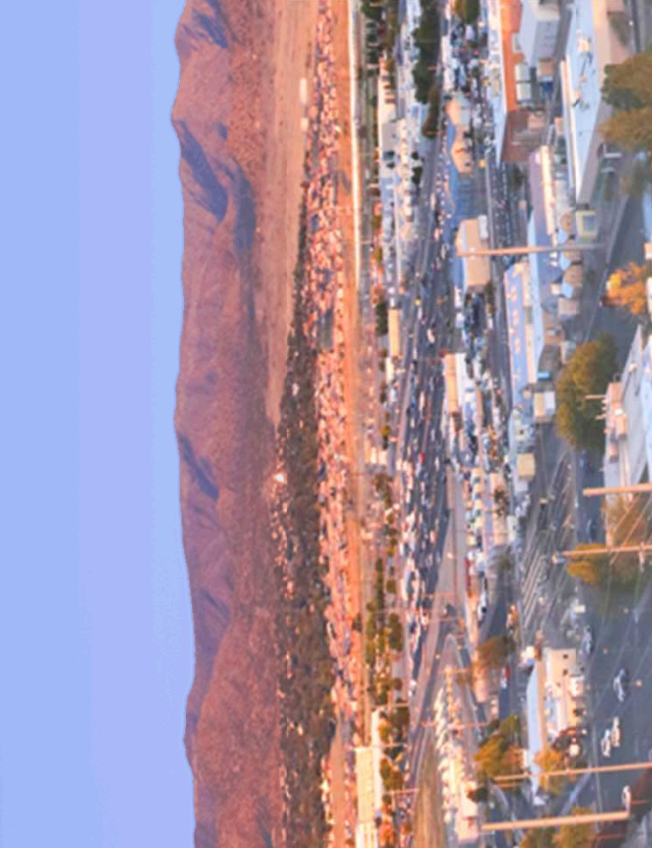


LCO is a non-ideal solution, gradients not as dominant as some believe

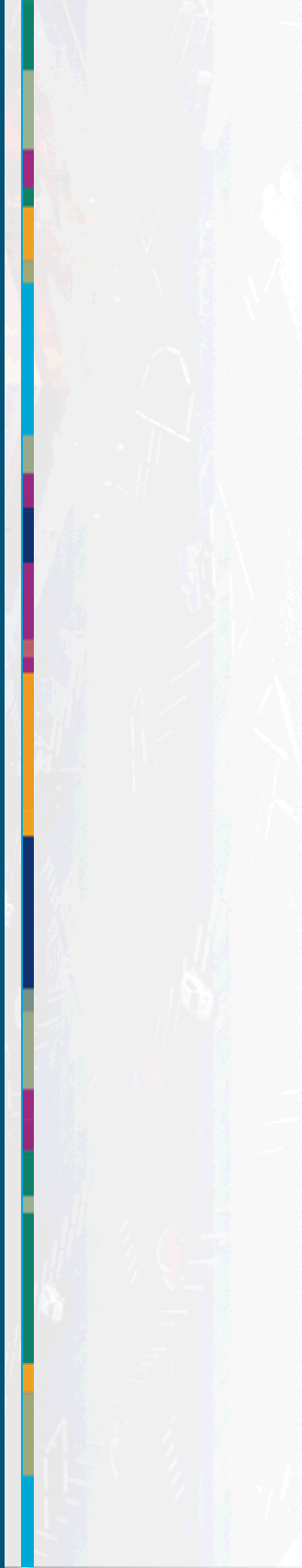


Separator/collector stiffness/boundary conditions influence electrode mesostructure evolution

Details at the mesoscale influence cell performance ... and vice versa!



Future directions in electrode mesoscale modeling



Future extensions of current work

Effective property calculation, correlation, and upscaling

- Effective properties for all 16 NMC data sets from Ebner (2013)
- Effective properties from DEM simulations, validation to experiment

Graphite anode mesoscale analysis (current student with P. Mukherjee)

- Recent image data from V. Wood group

Coupled electrochemical-mechanical simulations

- Study of binder fraction, calendaring, and discharge rate
- More appropriate boundary conditions (separator/collector stiffness/transport, V. Wood/C. Arnold)
- Multiple charge-discharge cycles
- Full-cell (cathode, separator, anode) analysis

Upscaling and integration with cell-level codes (i.e. CAEBAT/OAS/VIBE)

Automated image-to-simulation – machine learning (LDRD just funded)

Silicon anode mesoscale analysis

- Inelastic binder mechanics (viscoelasticity, delamination, etc.)
- Large deformation multi-physics framework

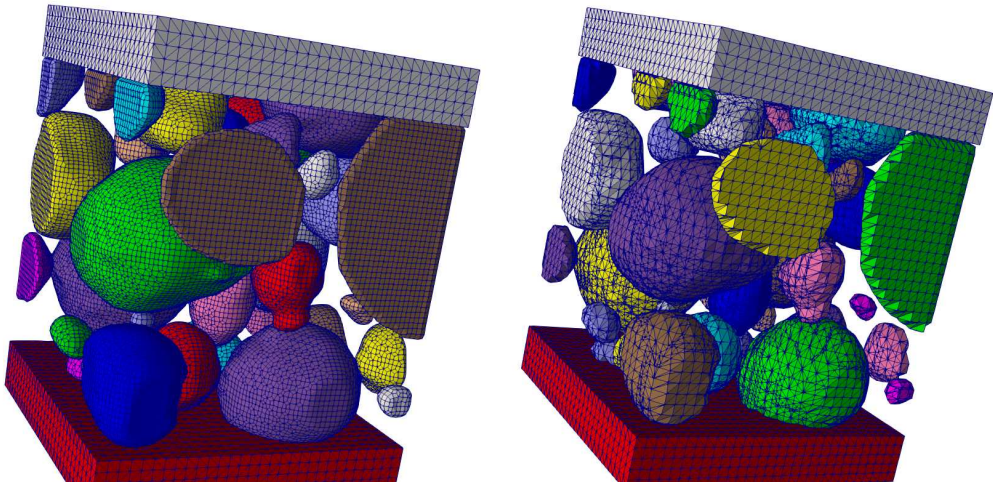
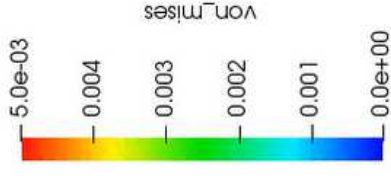
Application to 3D printed batteries, lithium plating and growth (2 current proposals to EERE/VTO)

There's still much work to be done in the areas discussed here

Large deformation multi-physics formulation

Concept:

- Solid mechanics c
- Transport/electr
- Transfer displacer



Demonstrated two-way coupling between mechanics and transport codes; developing robust particle contact

References and acknowledgments

Publications

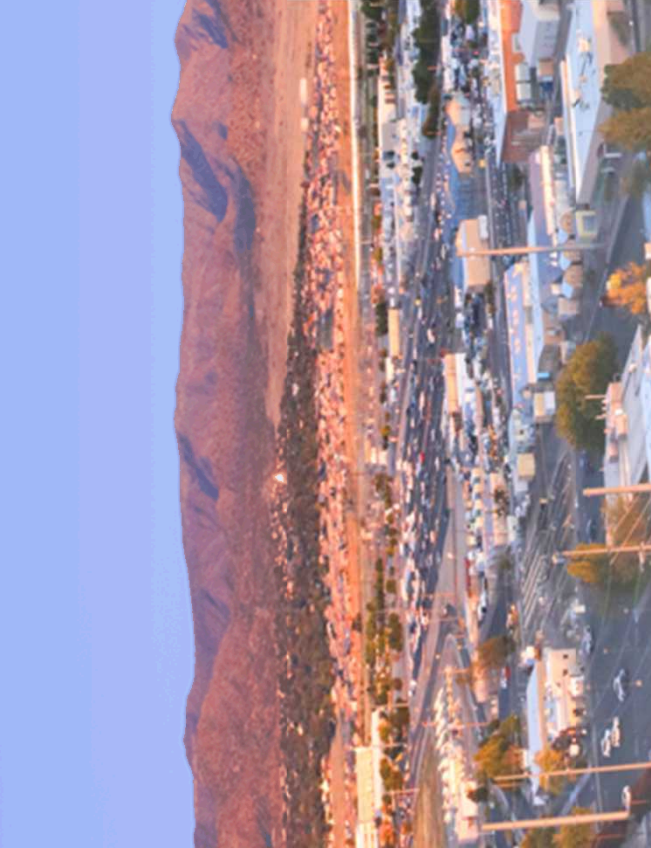
- Roberts *et al.*, “A Framework for Three-Dimensional Mesoscale Modeling of Anisotropic Swelling and Mechanical Deformation in Lithium-Ion Electrodes”, J. Electrochem. Soc. (2015) 10.1149/2.0081411jes
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Data

- V. Wood Group – ETH Zurich
- S. Thiele Group – U. Freiburg
- L. Zhu Group – IUPUI

Funding

- Sandia’s Laboratory Directed Research and Development (LDRD) program – 2013-2016
- Computer Aided Engineering for Batteries (CAEBAT) program, DOE/EERE/VTO – 2015-2018



Questions?

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