



## Modeling Perovskite Oxide Thermodynamics via the Compound Energy Formalism for Solar Thermochemistry

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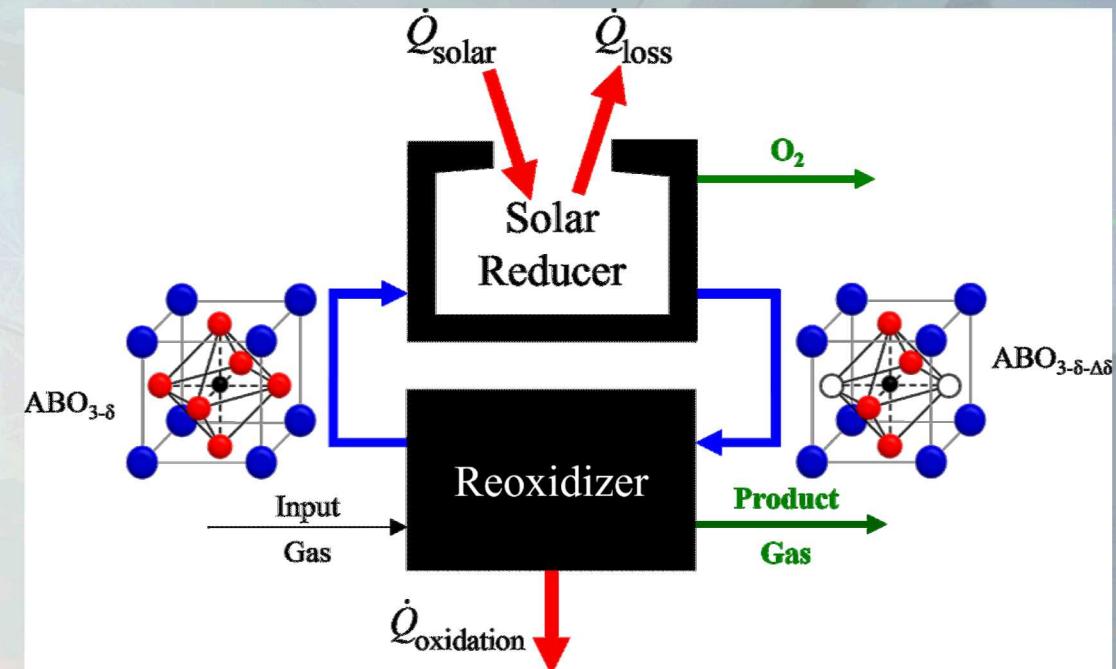
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## Motivation

- Mixed ion-electron conducting perovskites  $\text{ABO}_{3-\delta}$  are appealing for two-step thermochemical cycles
  - Continuous reducibility
  - Structural stability
  - Rapid kinetics
  - Large thermodynamic space via A/B-site substitution
- Thermodynamic modeling permits cycle performance prediction
  - Reduction-oxidation (redox) enthalpy and entropy
  - Predictions of  $\delta = f(T, p_{\text{O}_2})$
  - Fitting terms for substituents would allow mapping solution space



### Partial Substitution:

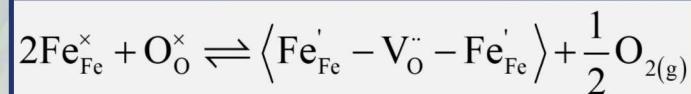
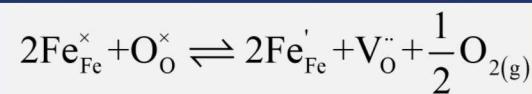
$\rightarrow (\text{A}'_x \text{A}_{1-x})\text{BO}_{3-\delta}$   
 $\text{A}(\text{B}'_x \text{B}_{1-x})\text{O}_{3-\delta}$   
 $(\text{A}'_x \text{A}_{1-x}) (\text{B}'_x \text{B}_{1-x})\text{O}_{3-\delta}$



## Strontium Ferrite ( $\text{SrFeO}_{3-\delta}$ )

- Promising candidate for thermochemistry
  - Thermodynamically stable
  - Low enthalpy of reduction
  - High reducibility
- Relevant applications:
  - Chemical looping  $\text{CH}_4$  combustion
  - Solid oxide fuel cells
  - Air separation
- Screening identified  $\text{A}' = \text{La, Ba}$  for air separation:
  - Improved reducibility
  - Similarly rapid kinetics

- Previous methods: lattice defect models



- Model fidelity varies with  $\delta, T, p_{\text{O}_2}$
- Doesn't readily capture multiple lattice phenomena
- Not readily extended to capture varying substitution

**Goal:** Model thermodynamics as a function of  $T, p_{\text{O}_2}$  and quantitatively capture substituent type and fraction influences



# Compound Energy Formalism (CEF)

- $(\text{Sr}^{2+}, \text{Ba}^{2+}, \text{La}^{3+})(\text{Fe}^{4+}, \text{Fe}^{3+})(\text{O}^{2-}, \text{V}_0)_3$ 
  - Three sublattices
- Semi-empirical: fitted, weighted by site fractions  $y_k$
- Captures redox without assuming single defect reactions
- Three components:
  1. Weighted sum of end members with  $T$  fit
    - $G_i$  represented as solid solution of materials with known thermodynamics, *e.g.*  $\text{Fe}_2\text{O}_3$ ,  $\text{SrO}$ ,  $\text{O}_2$
  2. Ideal mixing entropy, weighted by stoichiometry  $v_m$
  3. Excess terms (sublattice interactions) using Redlich-Kister polynomials

$$G^P = \sum_i \left( \overset{\circ}{G}_i \prod_{k \in S_i} y_k \right) + RT \sum_m \left( \nu_m \sum_{k \in S_m} y_k \ln y_k \right) + \overset{e}{G}^P$$

$${}^\circ G_{\text{oxide}} = {}^\circ G_{\text{end}} + {}^\circ G_{\text{mix}} + {}^\circ G_{\text{exs}}$$

$${}^E G^P = \sum_m \left( \prod_{k \in S_i} y_k \sum_n y_{k' \in S_m} \circ L_i^n \left[ y_{k' \in S_m} - y_{k \in S_m} \right]^n \right)$$



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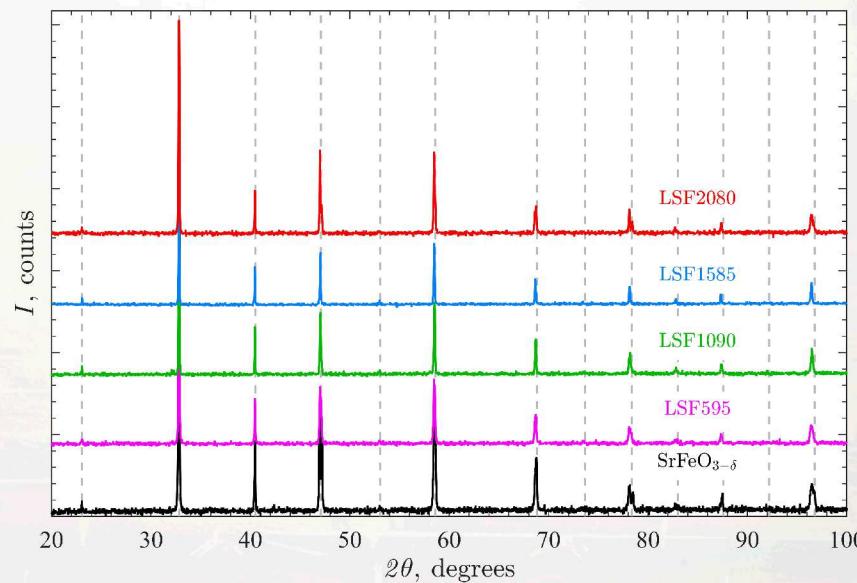
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## Synthesis & Characterization

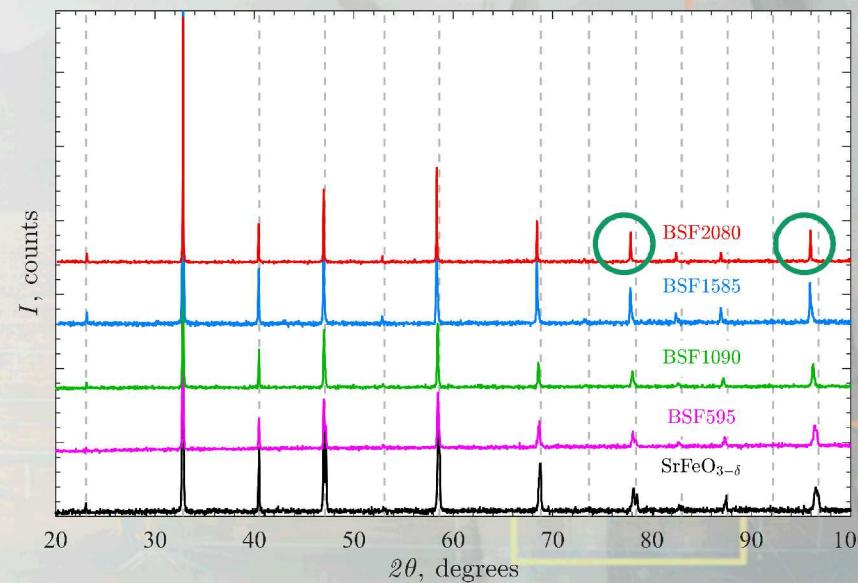
- Sol-gel method

- Metal nitrate precursors
- Calcination at 1250-1350 °C



- XRD

- Single phase, cubic or tetragonal SrFeO<sub>3-δ</sub>
- Peak shifting from Ba<sup>2+</sup> ionic radii





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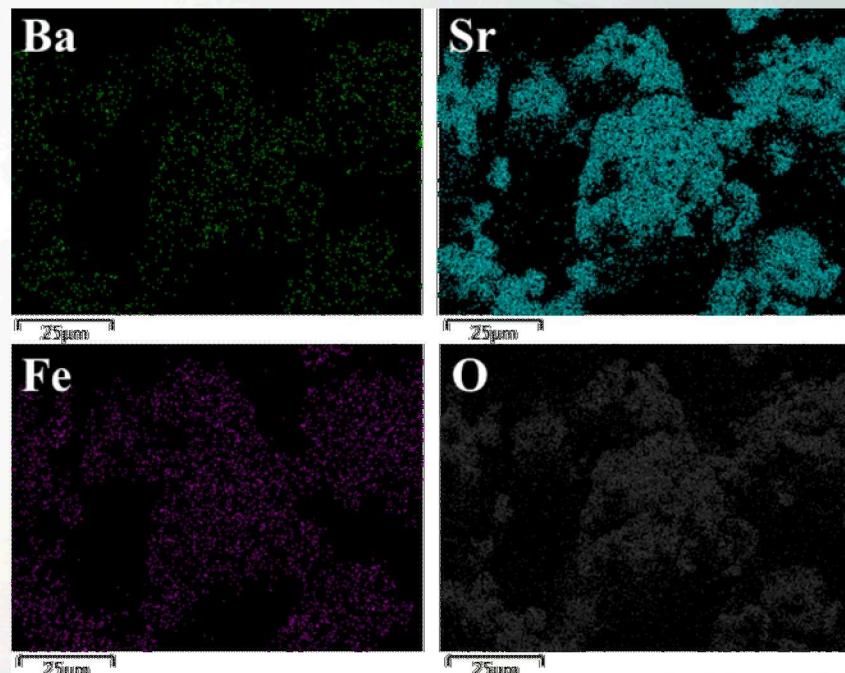
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## Synthesis & Characterization

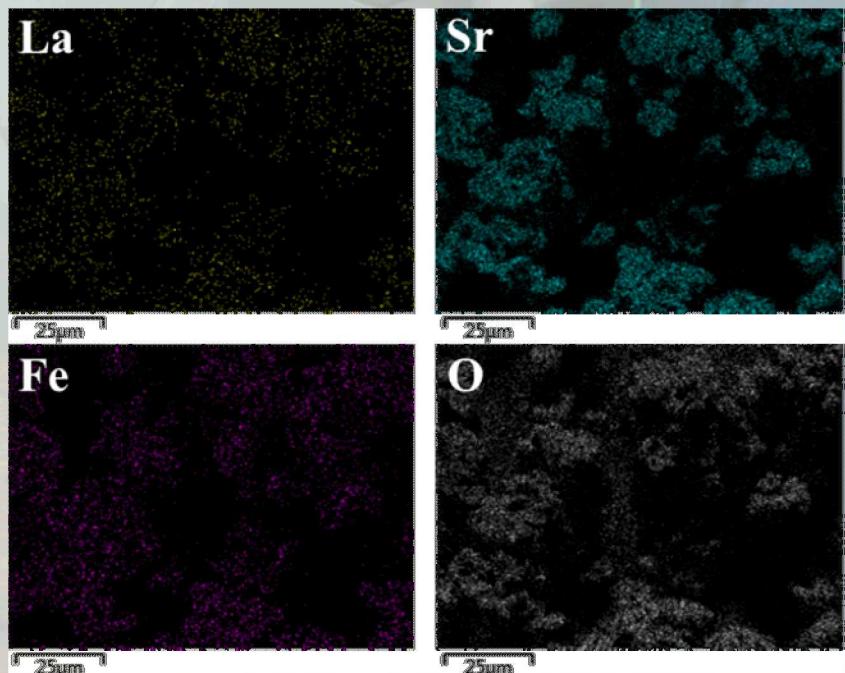
### • SEM/EDS

- Identified Sr, Fe, O, Ba, and La ions
- Homogenous cationic distributions
- No detected impurities
- No evidence of phase segregation

BSF



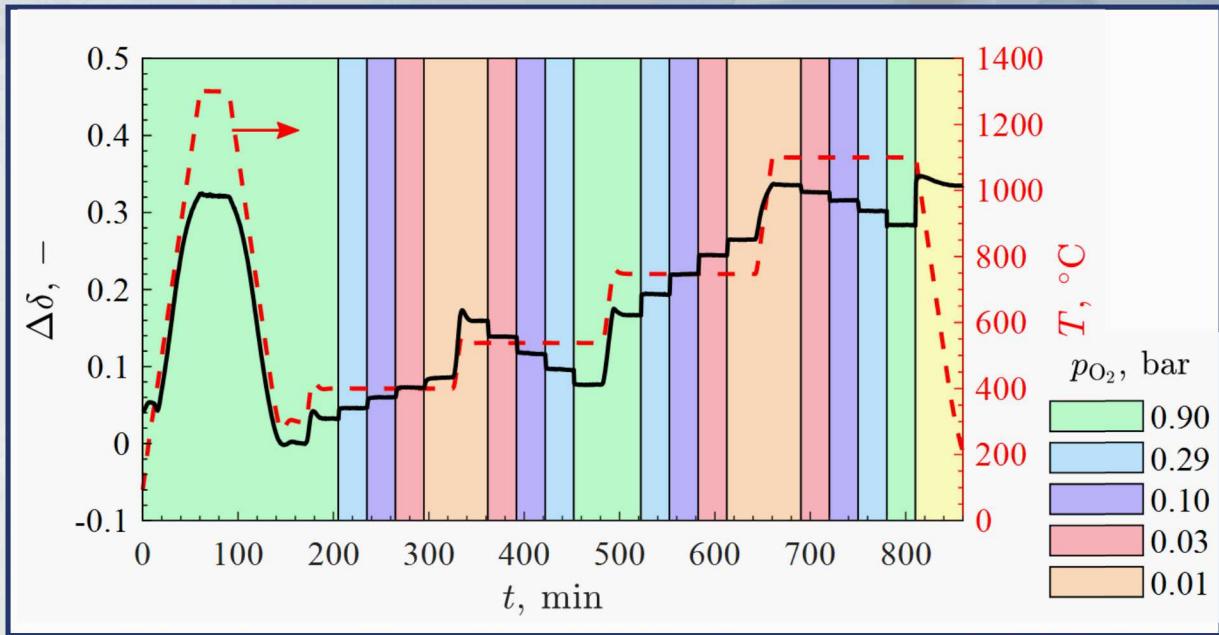
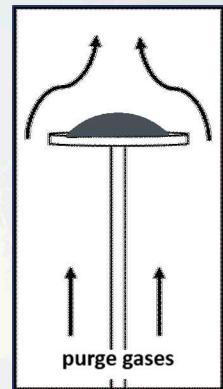
LSF





## TGA Equilibria

- Thermogravimetric analysis to measure equilibrium nonstoichiometry
  - 30 min ( $p_{O_2}$ ,  $T$ ) steps
- Confidence intervals calculated via error propagation of experimental variability
  - Two replicates performed per sample
  - Ten replicates performed for blank run
- $\delta_0 = \delta$  at  $p_{O_2} = 0.9$  bar,  $T = 300$  °C



### Naming convention:

BSF595 =  $Ba_{0.05}Sr_{0.95}FeO_{3-\delta}$   
BSF1090 =  $Ba_{0.1}Sr_{0.9}FeO_{3-\delta}$   
BSF1585 =  $Ba_{0.15}Sr_{0.85}FeO_{3-\delta}$   
BSF2080 =  $Ba_{0.2}Sr_{0.8}FeO_{3-\delta}$

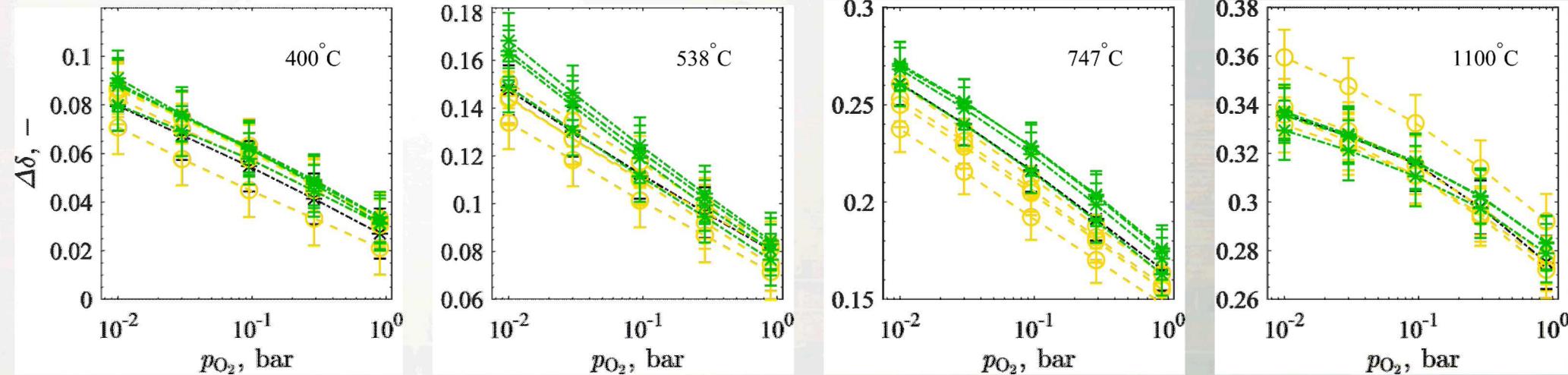
SF =  $SrFeO_{3-\delta}$

LSF595 =  $La_{0.05}Sr_{0.95}FeO_{3-\delta}$   
LSF1090 =  $La_{0.1}Sr_{0.9}FeO_{3-\delta}$   
LSF1585 =  $La_{0.15}Sr_{0.85}FeO_{3-\delta}$   
LSF2080 =  $La_{0.2}Sr_{0.8}FeO_{3-\delta}$



## Experimental Results

- LSF generally less reducible than SF/BSF, but trend changed at high  $T$
- “Best” composition highly dependent upon cycle conditions
- Modeling necessary to optimize composition to application





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## Compound Energy Model

- Simultaneously fit to improve model statistics
- End member terms ( ${}^{\circ}G_{\text{end}}$ ):
  - Linear fits with  $T$  sufficient
  - Similar  $T$ -dependence for La, Ba terms
- Mixing entropy terms ( ${}^{\circ}G_{\text{mix}}$ ):
  - Ideal intra-sublattice mixing
  - No fitted parameters
- Excess terms ( ${}^{\circ}G_{\text{exs}}$ ):
  - A-site interactions insignificant
  - 2<sup>nd</sup> order terms important at high  $T$
  - $\text{Fe}^{4+}$ - $\text{Fe}^{3+}$ ,  $\text{O}^{2-}$ - $\text{V}_0$  terms capture vacancy ordering

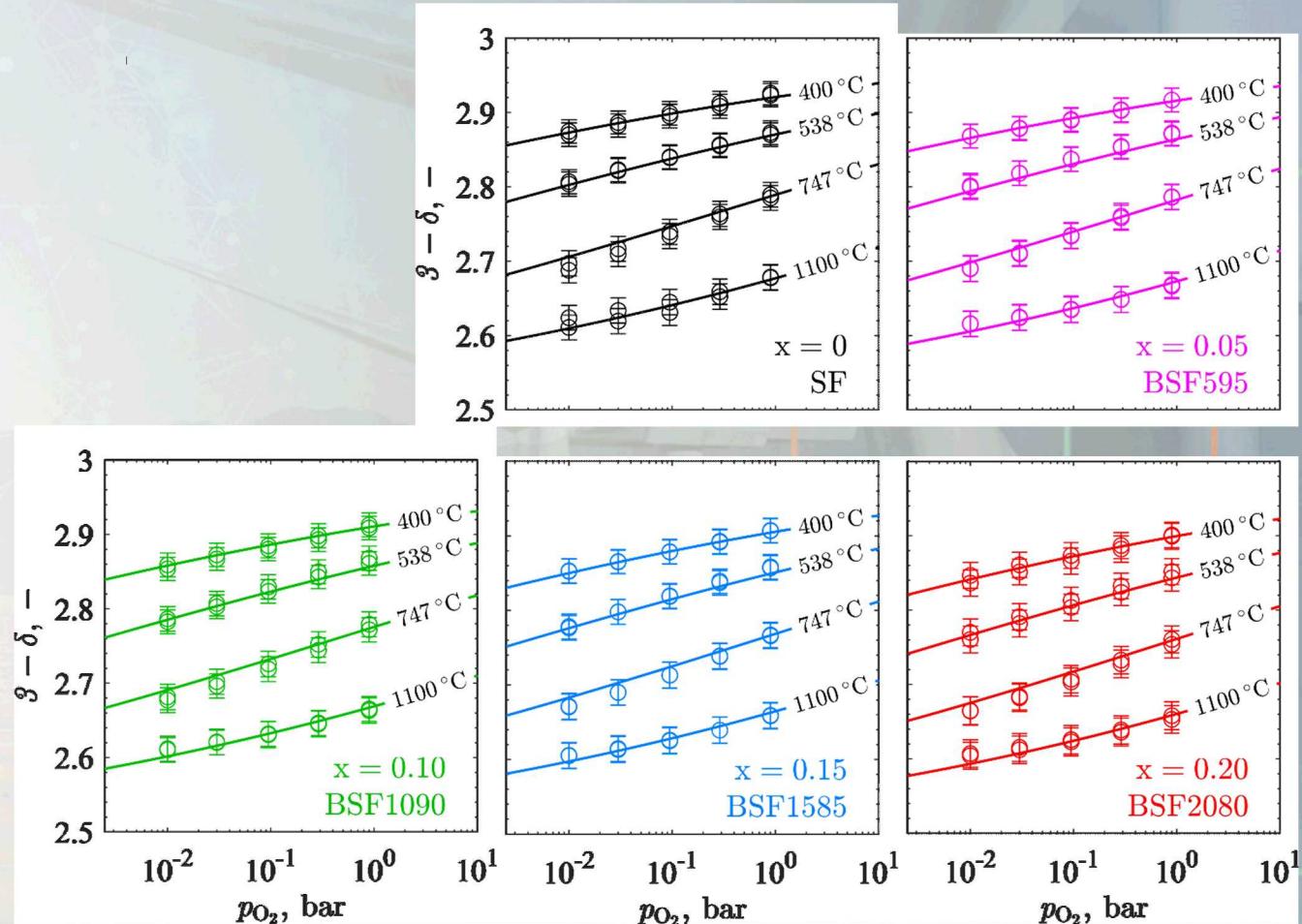
End Members	
$\text{Sr}^{2+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{Sr}^{2+} : \text{Fe}^{3+} : \text{O}^{2-}$
$\text{Ba}^{2+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{Ba}^{2+} : \text{Fe}^{3+} : \text{O}^{2-}$
$\text{La}^{3+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{La}^{3+} : \text{Fe}^{3+} : \text{O}^{2-}$
A : B : Vo (six terms)	
${}^{\circ}G_{\text{A:B:Vo}} = {}^{\circ}G_{\text{A:B:O}} - 3/2 {}^{\circ}G_{\text{O}_2\text{(g)}}$ (reciprocal relationship)	

Interaction	Gibbs Free Energy ( ${}^{\circ}G_{\text{exs}}$ ) Definition	
	0 <sup>th</sup> Order	1 <sup>st</sup> Order
$\text{Fe}^{3+} - \text{Fe}^{4+}$	$(7.67 \pm 0.09)\text{E-2 } T$ – $(1.15 \pm 0.71)\text{E-6 } T^2$	$(4.99 \pm 0.07)\text{E-1 } T$ + $(9.77 \pm 0.60)\text{E-5 } T^2$
$\text{O}^{2-} - \text{V}_0$	$-(2.24 \pm 0.14)\text{E-2 } T$ + $(9.06 \pm 1.32)\text{E-6 } T^2$	–



## CEF Model: BSF

- High experimental repeatability
  - Large overlap of marker error bounds
- All compositions within error bounds
- Interactions important at high  $T$ 
  - $T^2$  terms necessary at 1100 °C

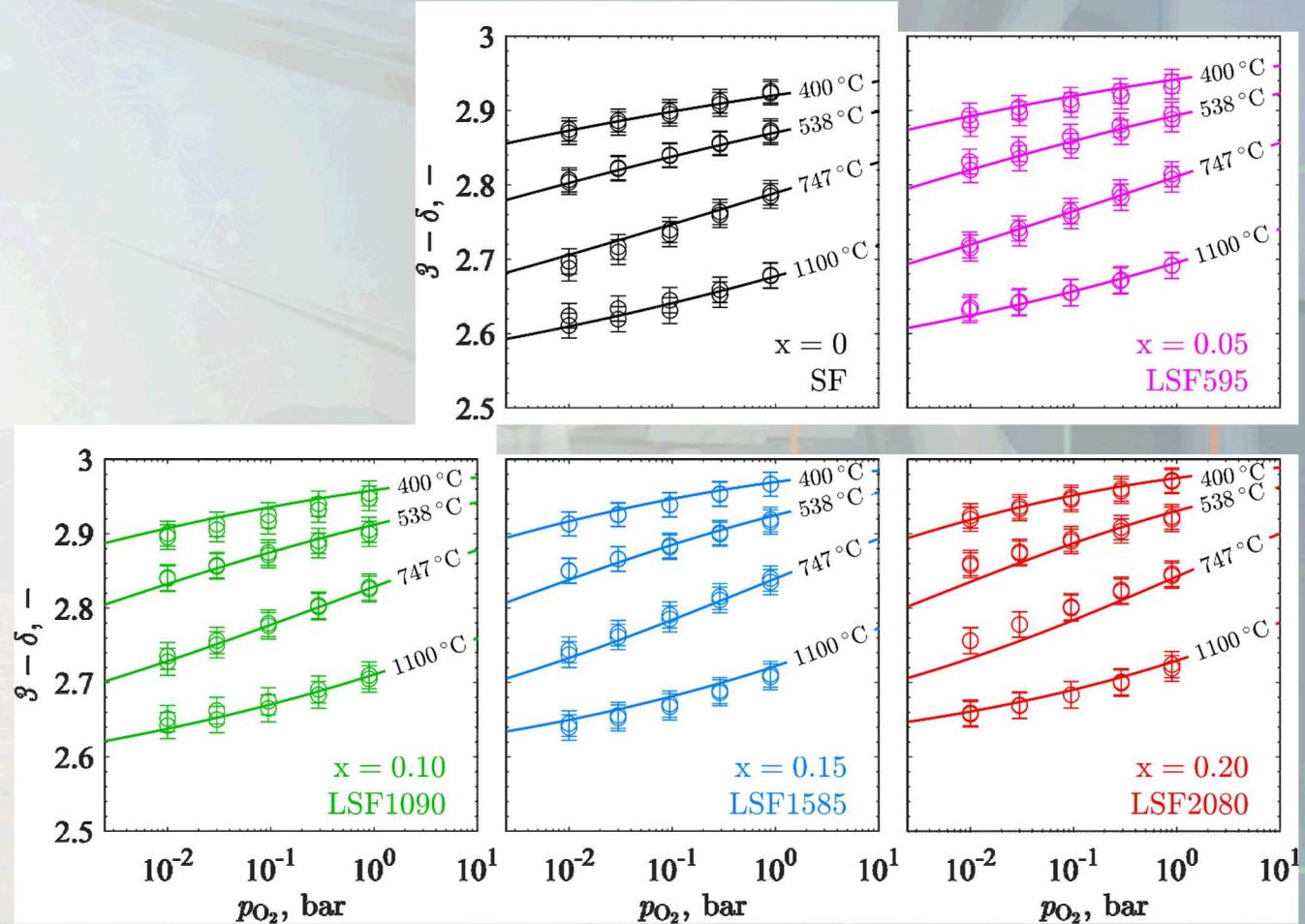




## CEF Model: LSF

- High experimental/model fidelity
  - LSF2080 largest error: low  $p_{O_2}$ , mid  $T$
- Interactions important at high  $T$
- Smaller  $\delta_0$  for LSF compared to BSF

x	BSF	LSF
0	$0.05 \pm 0.01$	
0.05	$0.05 \pm 0.01$	$0.04 \pm 0.02$
0.10	$0.06 \pm 0.01$	$0.02 \pm 0.01$
0.15	$0.06 \pm 0.01$	$0.00 \pm 0.00$
0.20	$0.07 \pm 0.01$	$0.01 \pm 0.01$





## Thermodynamics

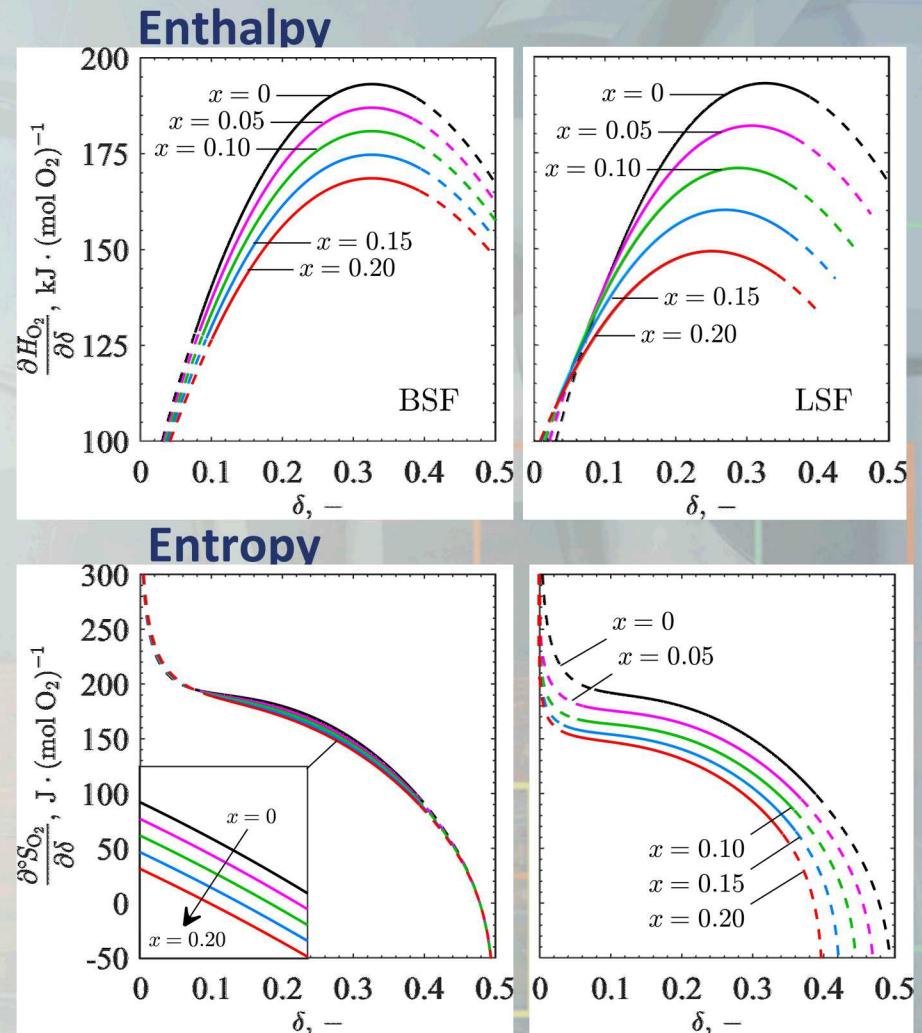
- Van't Hoff estimates of  $\Delta H_\delta$  and  $\Delta S_\delta$  from CEF fits
- Ba and La-substitution decreased  $\Delta H_\delta$
- Small (insignificant)  $\Delta S_\delta$  difference for BSF
- Larger LSF  $\Delta S_\delta$  difference: La<sup>3+</sup>/Fe<sup>3+</sup> charge compensation
  - Diminished reduction favorability ( $G = H - TS$ )

### BSF

Sample	$\Delta H_\delta$ kJ/mol O <sub>2</sub>	$\Delta S_\delta$ J/mol O <sub>2</sub> -K
SrFeO <sub>3</sub>	166 ± 1	145 ± 6
BSF595	161 ± 2	144 ± 7
BSF1090	156 ± 1	143 ± 7
BSF1585	152 ± 1	141 ± 7
BSF2080	147 ± 1	140 ± 7

### LSF

Sample	$\Delta H_\delta$ kJ/mol O <sub>2</sub>	$\Delta S_\delta$ J/mol O <sub>2</sub> -K
SrFeO <sub>3</sub>	166 ± 1	145 ± 6
LSF595	159 ± 1	133 ± 5
LSF1090	151 ± 1	124 ± 4
LSF1585	142 ± 1	118 ± 6
LSF2080	133 ± 0	114 ± 3





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## Summary

- CEF captured substituted strontium ferrite redox behavior as a function of temperature,  $O_2$  pressure, and Ba/La substituent fraction
  - Ba substitution more appealing for lower temperature cycles
  - La substitution more appealing for higher temperature cycles
- Provides framework for comparing substituents with general model
- Thermodynamics may be coupled to cycle model for process optimization



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