



Modeling Perovskite Oxide Thermodynamics via the Compound Energy Formalism for Solar Thermochemistry

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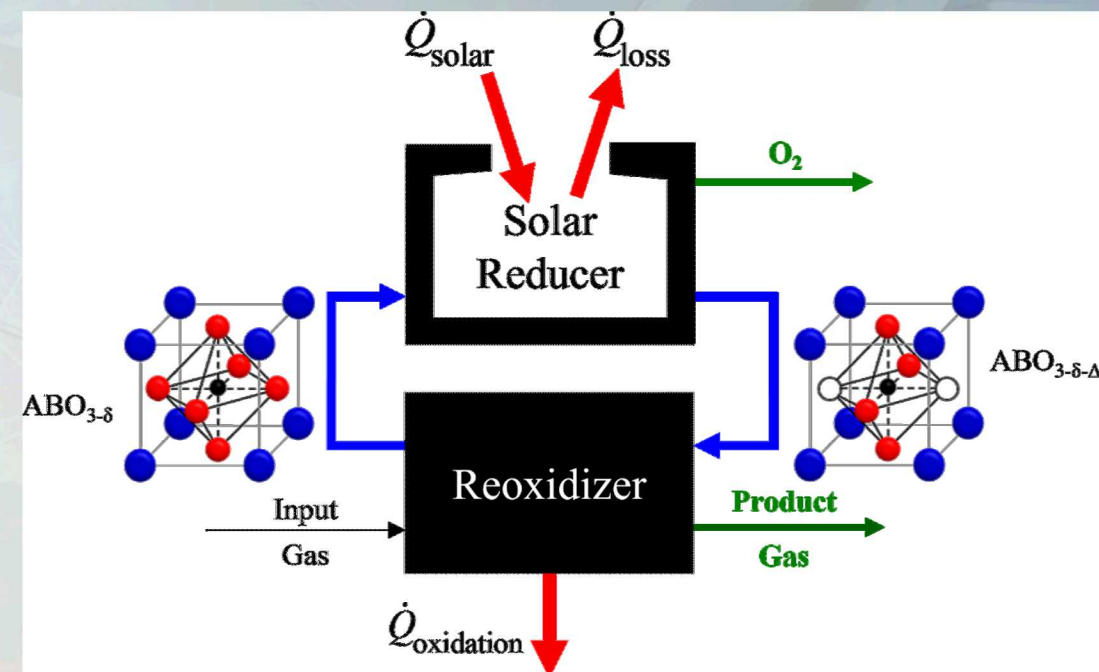
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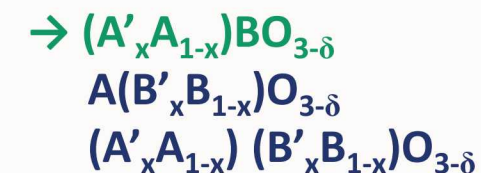


Motivation

- Mixed ion-electron conducting perovskites $ABO_{3-\delta}$ are appealing for two-step thermochemical cycles
 - Continuous reducibility
 - Structural stability
 - Rapid kinetics
 - Large thermodynamic space via A/B-site substitution
- Thermodynamic modeling permits cycle performance prediction
 - Reduction-oxidation (redox) enthalpy and entropy
 - Predictions of $\delta = f(T, p_{O_2})$
 - Fitting terms for substituents would allow mapping solution space



Partial Substitution:





Strontium Ferrite ($\text{SrFeO}_{3-\delta}$)

- Promising candidate for thermochemistry

- Thermodynamically stable
- Low enthalpy of reduction
- High reducibility

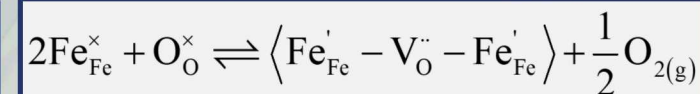
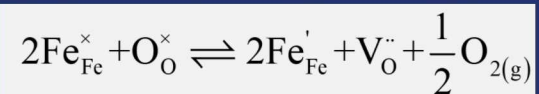
- Relevant applications:

- Chemical looping CH_4 combustion
- Solid oxide fuel cells
- Air separation

- Screening identified $\text{A}' = \text{La}, \text{Ba}$ for air separation:

- Improved reducibility
- Similarly rapid kinetics

- Previous methods: lattice defect models



- Model fidelity varies with $\delta, T, p_{\text{O}_2}$
- Doesn't readily capture multiple lattice phenomena
- Not readily extended to capture varying substitution

Goal: Model thermodynamics as a function of T, p_{O_2} and quantitatively capture substituent type and fraction influences



Compound Energy Formalism (CEF)

- $(\text{Sr}^{2+}, \text{Ba}^{2+}, \text{La}^{3+})(\text{Fe}^{4+}, \text{Fe}^{3+})(\text{O}^{2-}, \text{V}_\text{O})_3$
 - Three sublattices
- Semi-empirical: fitted, weighted by site fractions y_k
- Captures redox without assuming single defect reactions
- Three components:
 1. Weighted sum of end members with T fit
 ${}^\circ G_i$ represented as solid solution of materials with known thermodynamics, *e.g.* Fe_2O_3 , SrO , O_2
 2. Ideal mixing entropy, weighted by stoichiometry ν_m
 3. Excess terms (sublattice interactions) using Redlich-Kister polynomials

$$G^P = \sum_i \left({}^\circ G_i \prod_{k \in S_i} y_k \right) + RT \sum_m \left(\nu_m \sum_{k \in S_m} y_k \ln y_k \right) + {}^E G^P$$

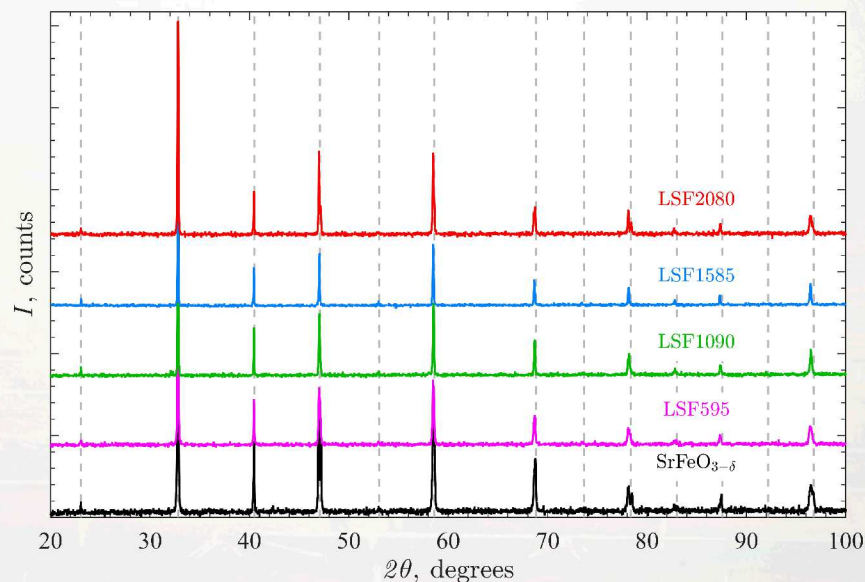
$${}^\circ G_{\text{oxide}} = {}^\circ G_{\text{end}} + {}^\circ G_{\text{mix}} + {}^\circ G_{\text{exs}}$$

$${}^E G^P = \sum_m \left(\prod_{k \in S_i} y_k \sum_n y_{k' \in S_m} {}^\circ L_i^n \left[y_{k' \in S_m} - y_{k \in S_m} \right]^n \right)$$

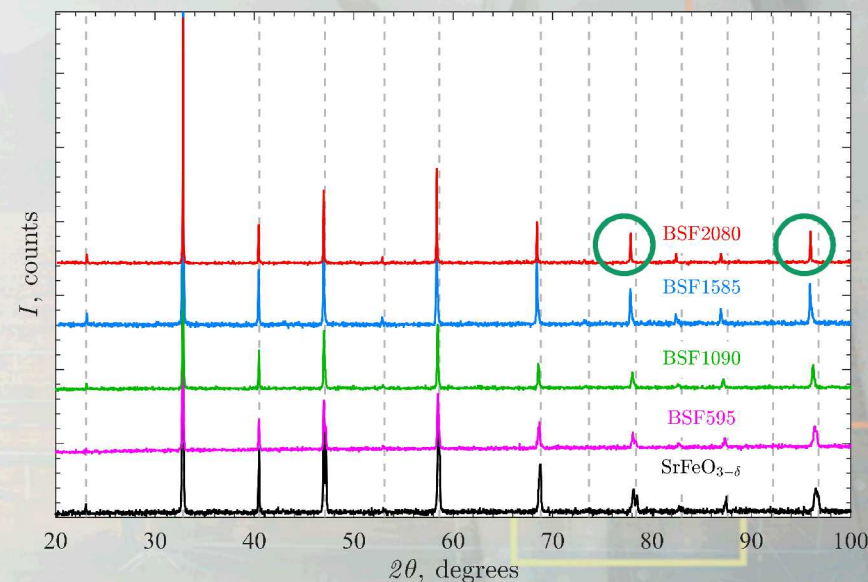


Synthesis & Characterization

- Sol-gel method
 - Metal nitrate precursors
 - Calcination at 1250-1350 °C



- XRD
 - Single phase, cubic or tetragonal $\text{SrFeO}_{3-\delta}$
 - Peak shifting from Ba^{2+} ionic radii



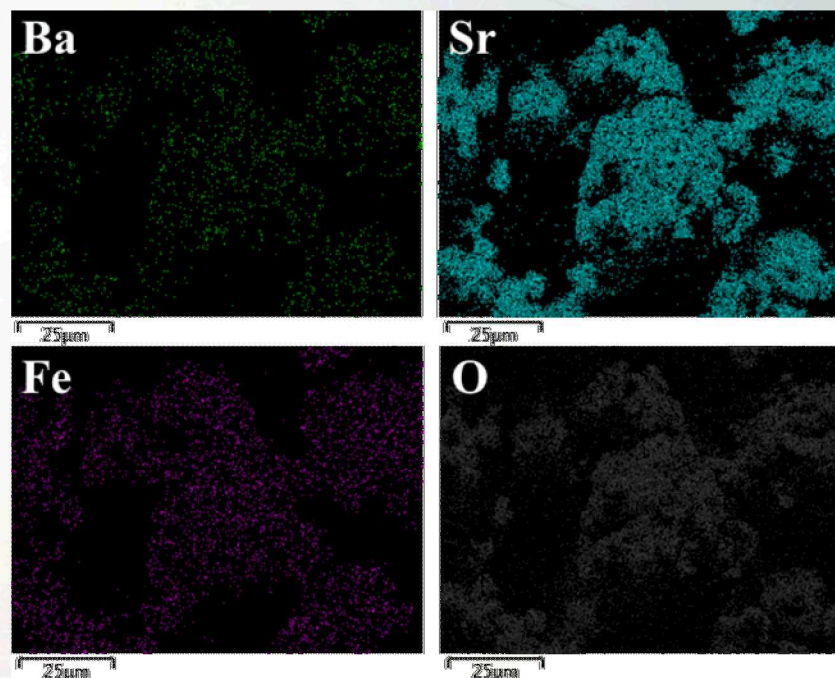


Synthesis & Characterization

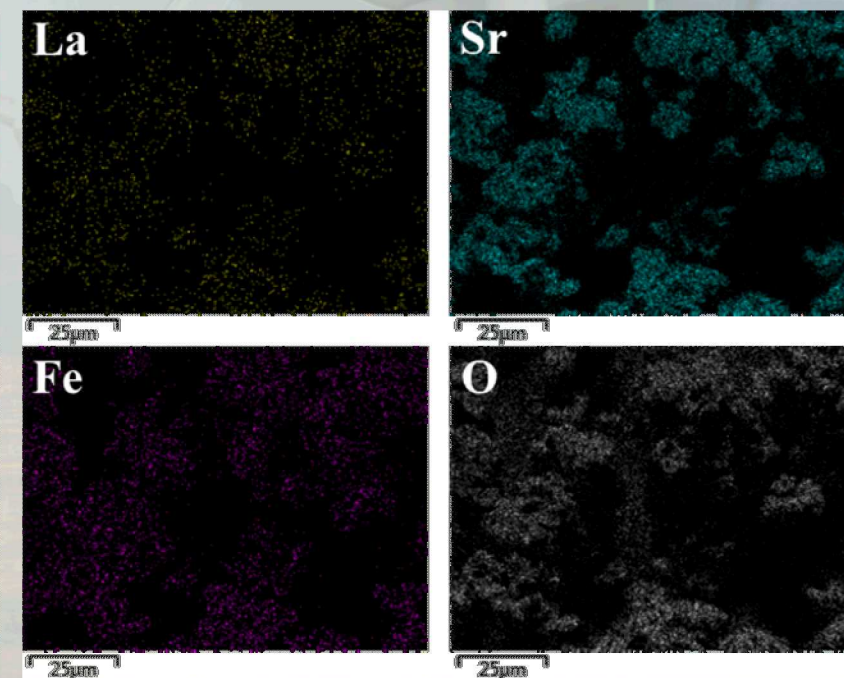
• SEM/EDS

- Identified Sr, Fe, O, Ba, and La ions
- Homogenous cationic distributions
- No detected impurities
- No evidence of phase segregation

BSF



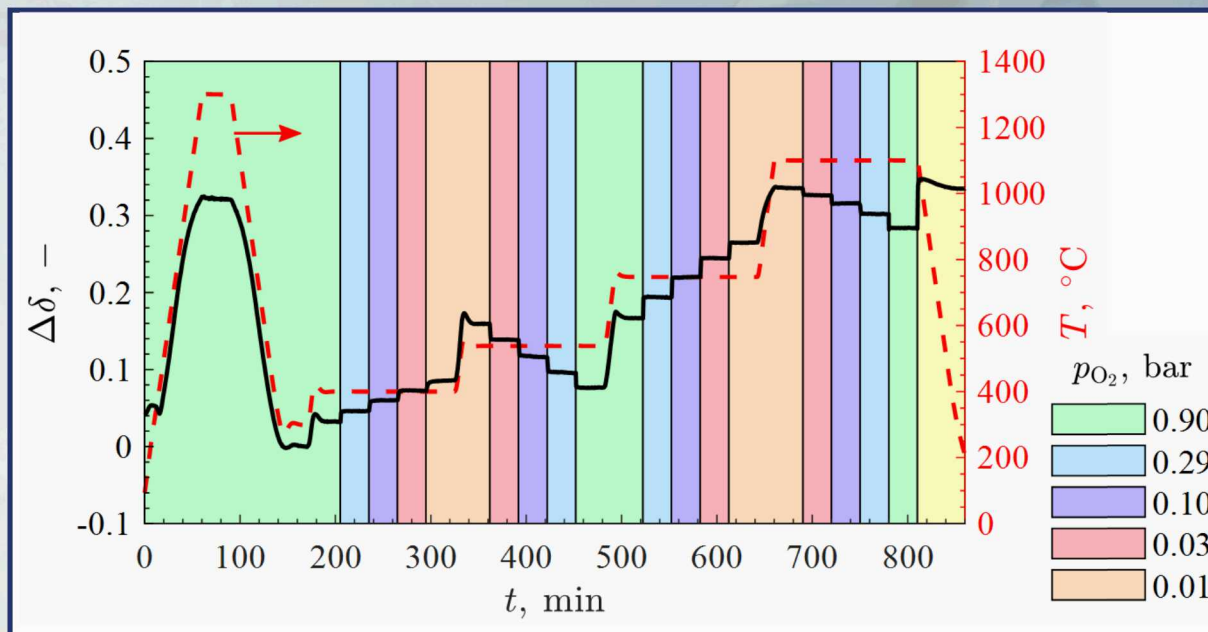
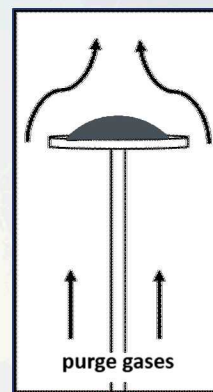
LSF





TGA Equilibria

- Thermogravimetric analysis to measure equilibrium nonstoichiometry
 - 30 min (p_{O_2} , T) steps
- Confidence intervals calculated via error propagation of experimental variability
 - Two replicates performed per sample
 - Ten replicates performed for blank run
- $\delta_0 = \delta$ at $p_{O_2} = 0.9$ bar, $T = 300$ °C



Naming convention:

BSF595 = $Ba_{0.05}Sr_{0.95}FeO_{3-\delta}$
BSF1090 = $Ba_{0.1}Sr_{0.9}FeO_{3-\delta}$
BSF1585 = $Ba_{0.15}Sr_{0.85}FeO_{3-\delta}$
BSF2080 = $Ba_{0.2}Sr_{0.8}FeO_{3-\delta}$

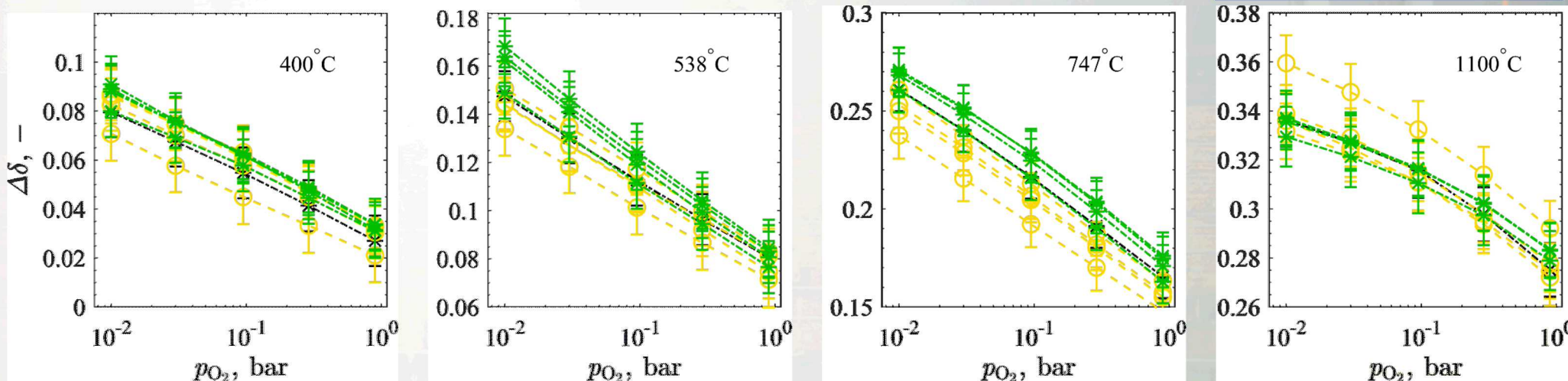
SF = $SrFeO_{3-\delta}$

LSF595 = $La_{0.05}Sr_{0.95}FeO_{3-\delta}$
LSF1090 = $La_{0.1}Sr_{0.9}FeO_{3-\delta}$
LSF1585 = $La_{0.15}Sr_{0.85}FeO_{3-\delta}$
LSF2080 = $La_{0.2}Sr_{0.8}FeO_{3-\delta}$



Experimental Results

- LSF generally less reducible than SF/BSF, but trend changed at high T
- “Best” composition highly dependent upon cycle conditions
- Modeling necessary to optimize composition to application





Compound Energy Model

- Simultaneously fit to improve model statistics
- End member terms ($^{\circ}G_{\text{end}}$):
 - Linear fits with T sufficient
 - Similar T -dependence for La, Ba terms
- Mixing entropy terms ($^{\circ}G_{\text{mix}}$):
 - Ideal intra-sublattice mixing
 - No fitted parameters
- Excess terms ($^{\circ}G_{\text{exs}}$):
 - A-site interactions insignificant
 - 2nd order terms important at high T
 - Fe^{4+} - Fe^{3+} , O^{2-} - V_0 terms capture vacancy ordering

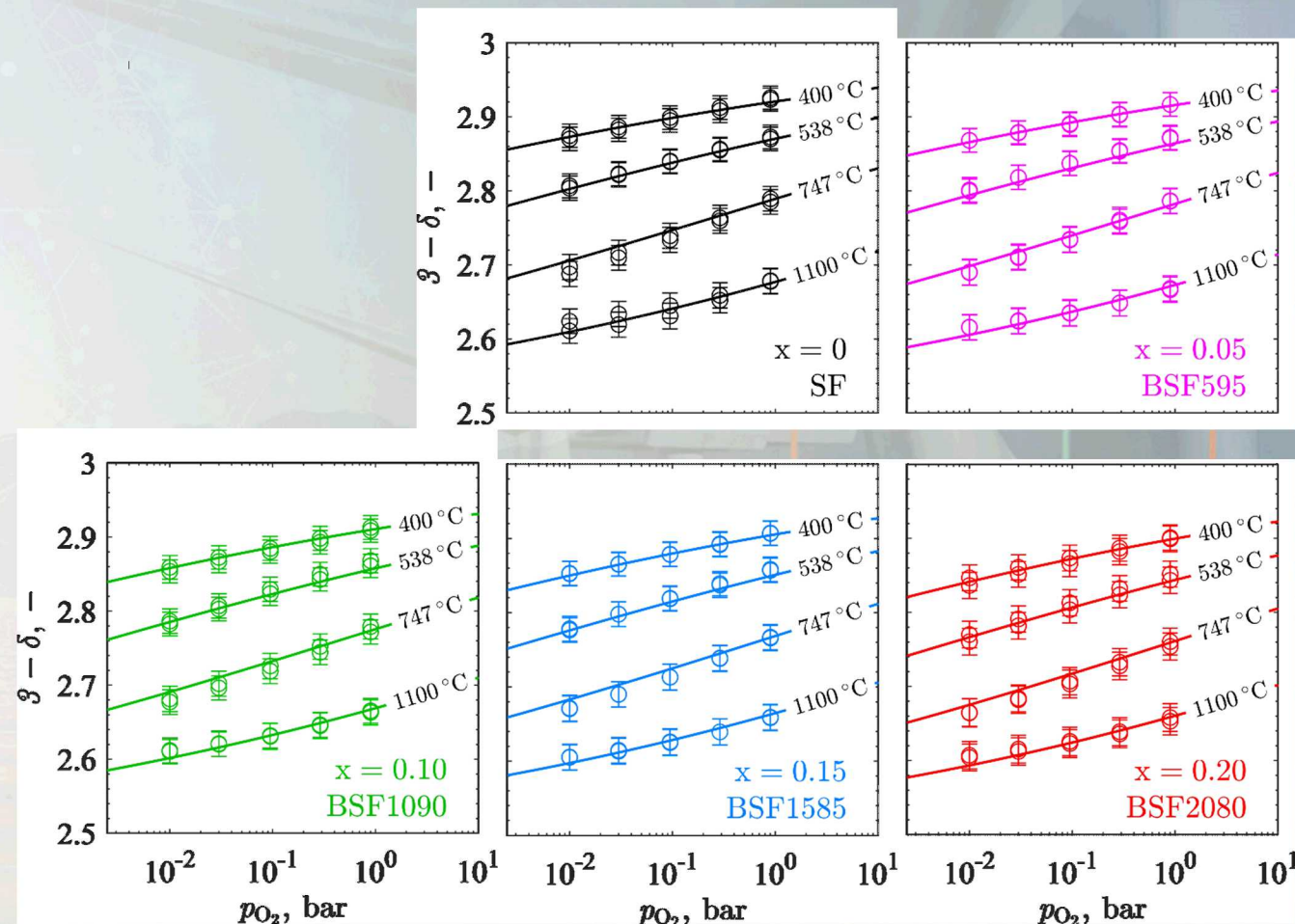
End Members	
$\text{Sr}^{2+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{Sr}^{2+} : \text{Fe}^{3+} : \text{O}^{2-}$
$\text{Ba}^{2+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{Ba}^{2+} : \text{Fe}^{3+} : \text{O}^{2-}$
$\text{La}^{3+} : \text{Fe}^{4+} : \text{O}^{2-}$	$\text{La}^{3+} : \text{Fe}^{3+} : \text{O}^{2-}$
A : B : Vo (six terms)	
$^{\circ}G_{\text{A:B:Vo}} = ^{\circ}G_{\text{A:B:O}} - 3/2^{\circ}G_{\text{O2(g)}} \text{ (reciprocal relationship)}$	

Interaction	Gibbs Free Energy ($^{\circ}G_{\text{exs}}$) Definition	
	0 th Order	1 st Order
$\text{Fe}^{3+} - \text{Fe}^{4+}$	$(7.67 \pm 0.09)\text{E-}2 \text{ } T$ $- (1.15 \pm 0.71)\text{E-}6 \text{ } T^2$	$(4.99 \pm 0.07)\text{E-}1 \text{ } T$ $+ (9.77 \pm 0.60)\text{E-}5 \text{ } T^2$
$\text{O}^{2-} - \text{V}_0$	$- (2.24 \pm 0.14)\text{E-}2 \text{ } T$ $+ (9.06 \pm 1.32)\text{E-}6 \text{ } T^2$	—



CEF Model: BSF

- High experimental repeatability
 - Large overlap of marker error bounds
- All compositions within error bounds
- Interactions important at high T
 - T^2 terms necessary at 1100 °C

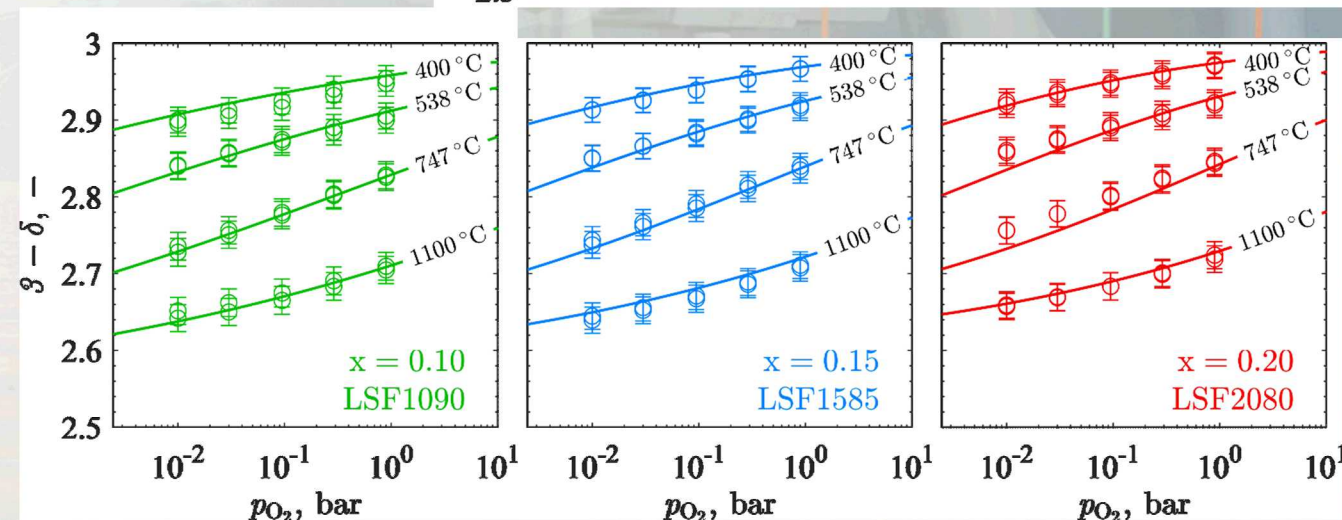
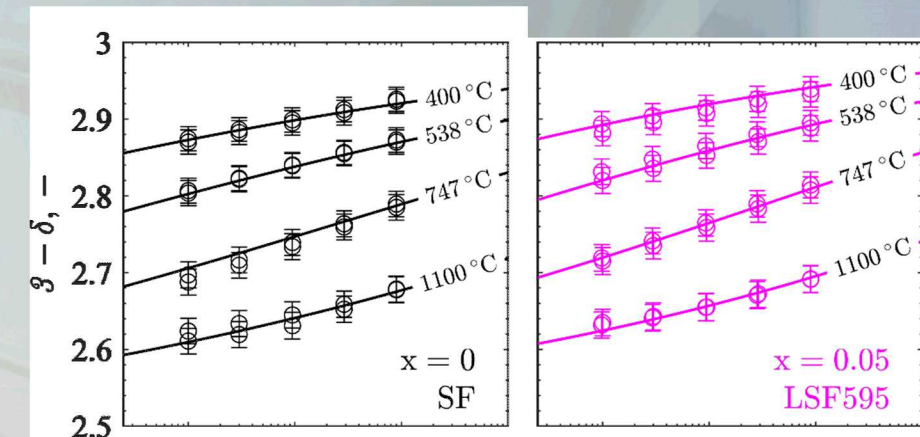




CEF Model: LSF

- High experimental/model fidelity
 - LSF2080 largest error: low p_{O_2} , mid T
- Interactions important at high T
- Smaller δ_0 for LSF compared to BSF

x	BSF	LSF
0	0.05 ± 0.01	
0.05	0.05 ± 0.01	0.04 ± 0.02
0.10	0.06 ± 0.01	0.02 ± 0.01
0.15	0.06 ± 0.01	0.00 ± 0.00
0.20	0.07 ± 0.01	0.01 ± 0.01





Thermodynamics

- Van't Hoff estimates of ΔH_δ and ΔS_δ from CEF fits
- Ba and La-substitution decreased ΔH_δ
- Small (insignificant) ΔS_δ difference for BSF
- Larger LSF ΔS_δ difference: $\text{La}^{3+}/\text{Fe}^{3+}$ charge compensation
 - Diminished reduction favorability ($G = H - TS$)

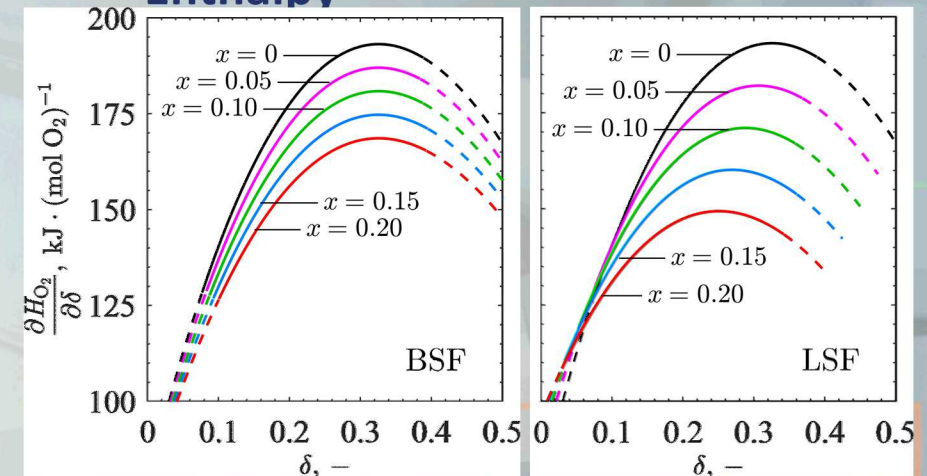
BSF

Sample	ΔH_δ kJ/mol O ₂	ΔS_δ J/mol O ₂ -K
SrFeO ₃	166 ± 1	145 ± 6
BSF595	161 ± 2	144 ± 7
BSF1090	156 ± 1	143 ± 7
BSF1585	152 ± 1	141 ± 7
BSF2080	147 ± 1	140 ± 7

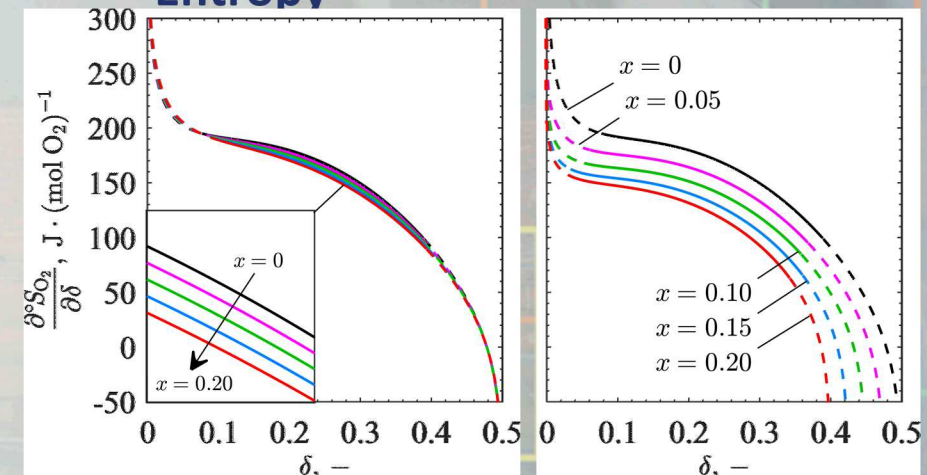
LSF

Sample	ΔH_δ kJ/mol O ₂	ΔS_δ J/mol O ₂ -K
SrFeO ₃	166 ± 1	145 ± 6
LSF595	159 ± 1	133 ± 5
LSF1090	151 ± 1	124 ± 4
LSF1585	142 ± 1	118 ± 6
LSF2080	133 ± 0	114 ± 3

Enthalpy



Entropy





Summary

- CEF captured substituted strontium ferrite redox behavior as a function of temperature, O_2 pressure, and Ba/La substituent fraction
 - Ba substitution more appealing for lower temperature cycles
 - La substitution more appealing for higher temperature cycles
- Provides framework for comparing substituents with general model
- Thermodynamics may be coupled to cycle model for process optimization



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