



Sandia  
National  
Laboratories

SAND2020-3586PE

# Addressing Complexity in Advanced Manufacturing

*PRESENTED BY*

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## Background

### Complexity is available

- machining
- additive manufacturing

### But, complexity isn't free

- uncertainties
- process-structure-properties
- metrology
- process monitoring & control

## Teaching

# Sandia National Laboratories



A National Security Science & Engineering Laboratory

- “Exceptional service in the national interest”

Nuclear Weapons

Defense Systems & Assessments

Energy & Climate

International, Homeland, & Nuclear Security





# 4 Complexity is Available

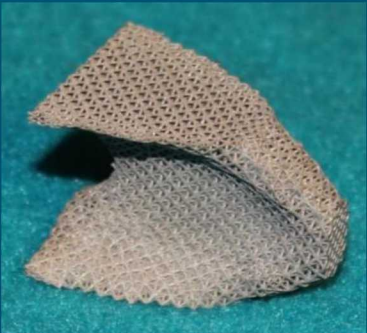
Advanced manufacturing provides ever expanding design freedom & value

- complex freeforms, internal structures, integration

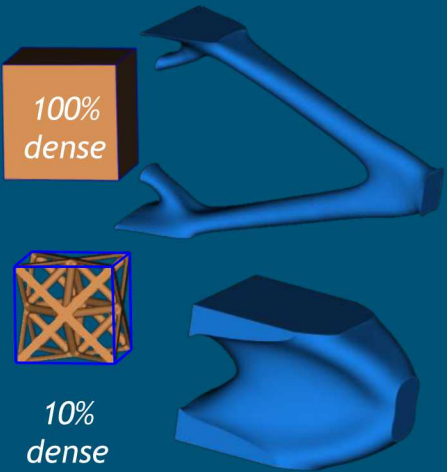
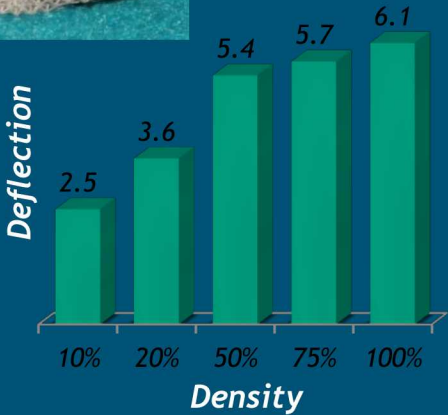
Geometry

Engineered materials

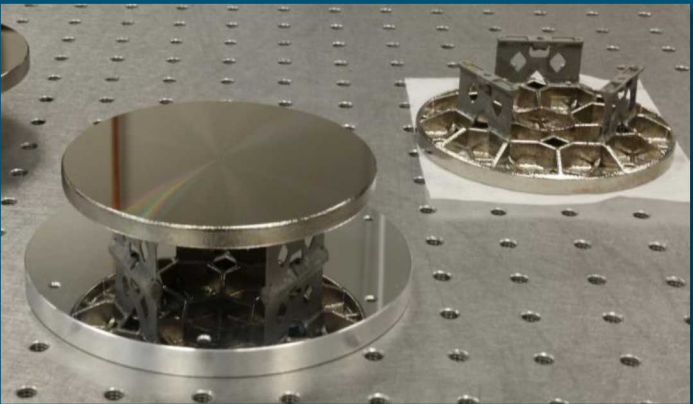
- gradients & microstructure
- multi-material integration
- “print everything inside the box, not just the box”



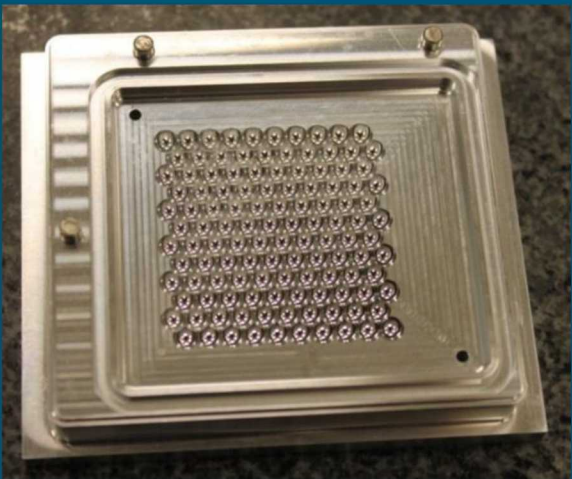
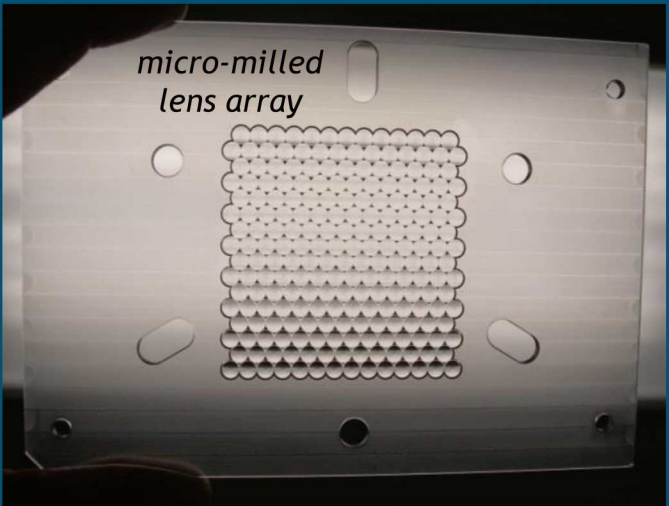
Jared, B., *Scripta Materialia*, 2017



Mitchell structure optimizing stiffness w/ mixed mass using PLATO  
Robbins, *Add Mfg*, 2016



AM Ti6Al4V mirror w/ diamond turned  
electroless Ni coating  
Jared, *ASPE*, 2017



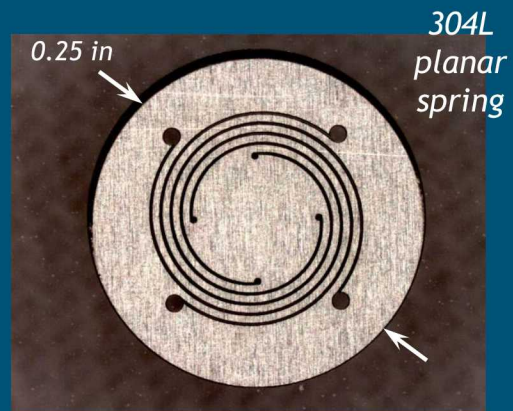
lens mold  
for  
casting  
micro-  
optic  
arrays



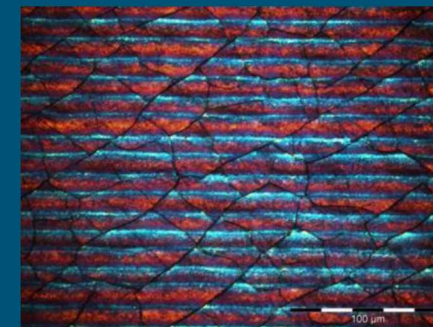
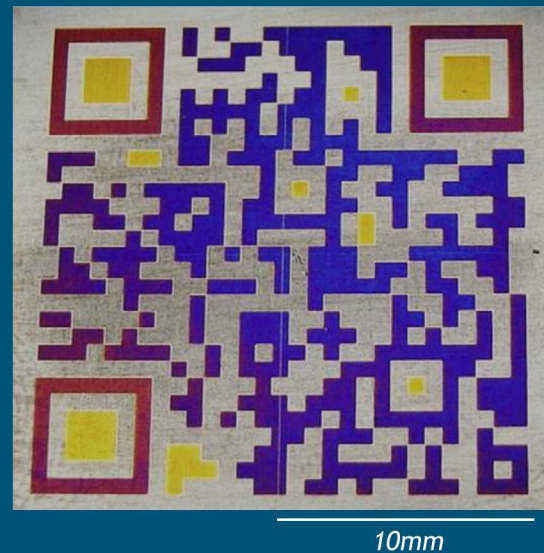
# Ultra-Fast Pulsed Laser Processing



fs cut 304L meso spring  
Jared, Int Con Micro Mfg, 2012

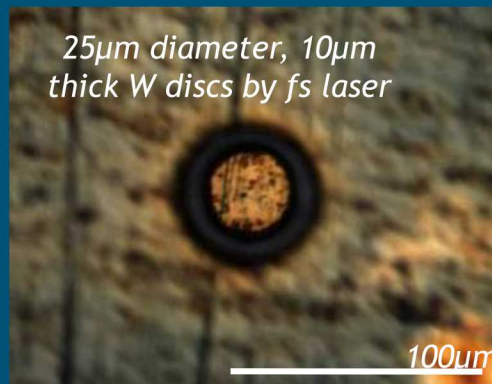


304L  
planar  
spring

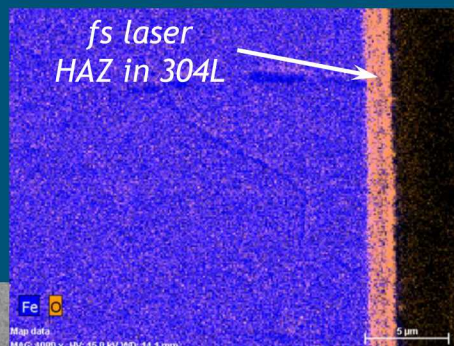


ns titanium oxides  
Adams, Surf & Coat Tech, 2014

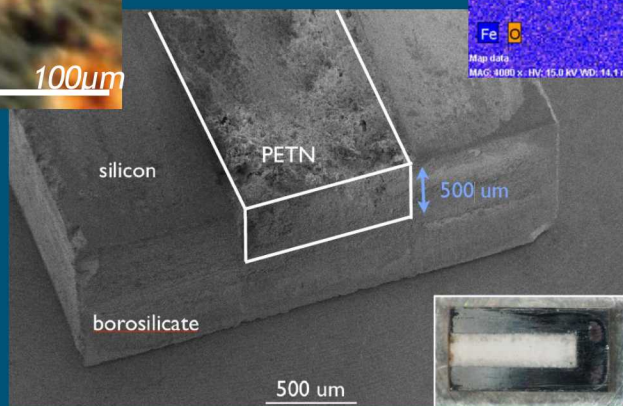
*surface modifications*



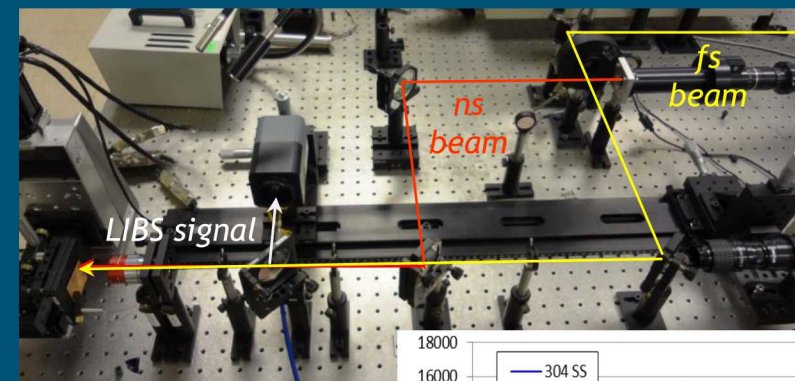
*meso scale  
features*



fs laser  
HAZ in 304L

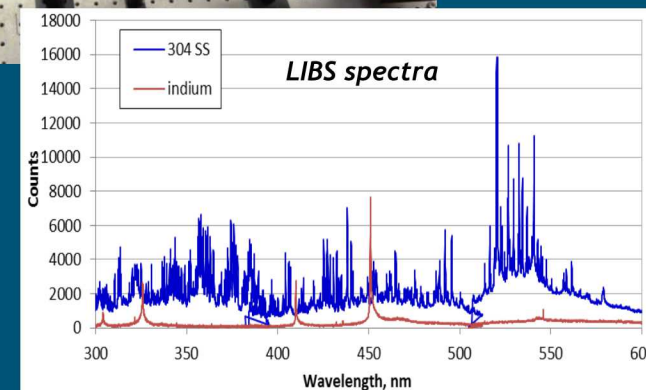


R. Wixom, Int Pyro Sem & Symp, 2008.



Jared,  
SAND2012-8653

*laser induced  
breakdown  
spectroscopy (LIBS)*

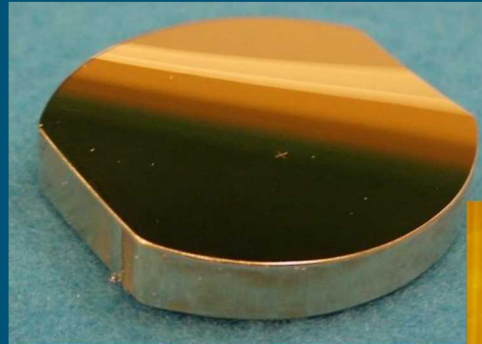




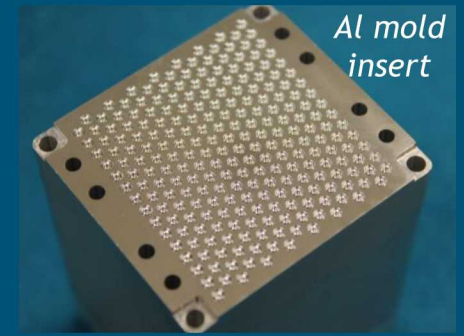
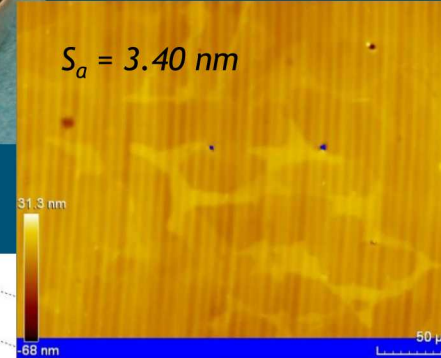
# Precision Machining



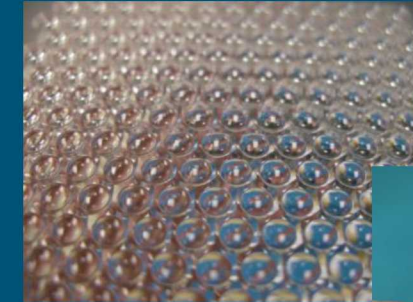
Yasda YMC-430  
micro milling



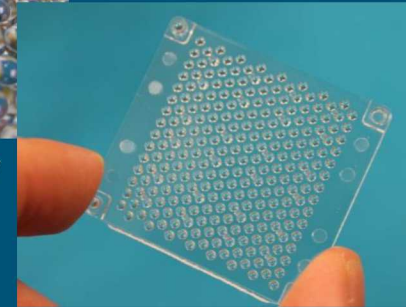
non-rotationally symmetric  
freeform mirror,  
Boye, Opt Eng, 2013



Al mold  
insert



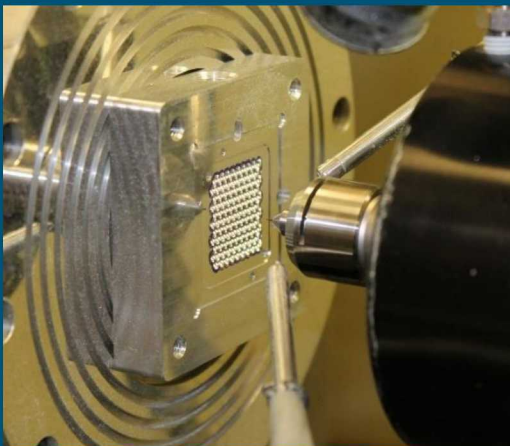
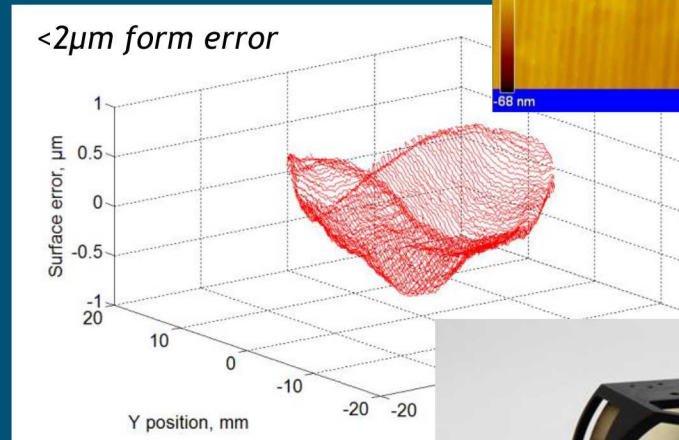
molded optic array



machined optic array



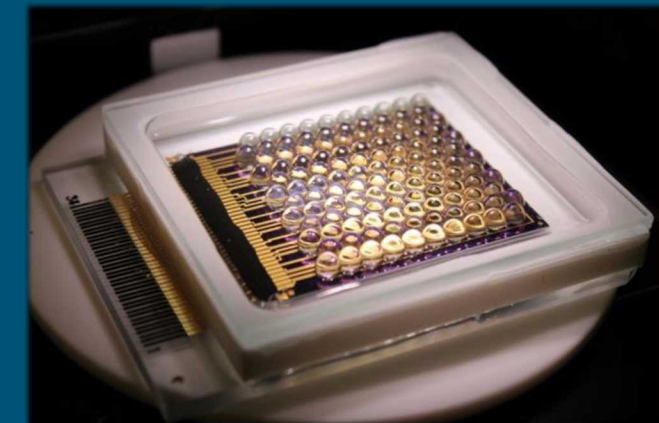
Moore 350FG  
diamond  
turning



diamond  
milling  
Al mold



four element monocular telescope



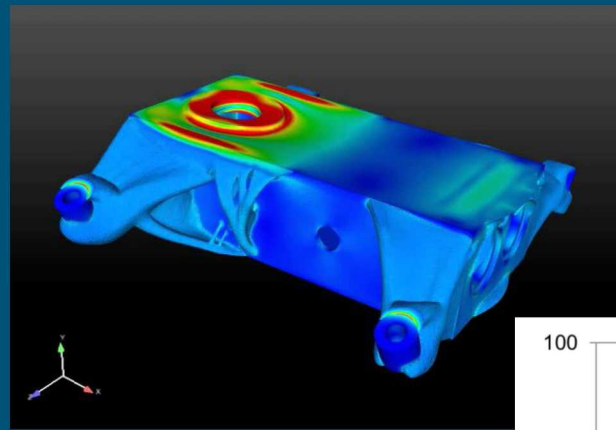
microsystems-enabled photovoltaics  
(MEPV) module  
Jared, Opt Exp, 2014



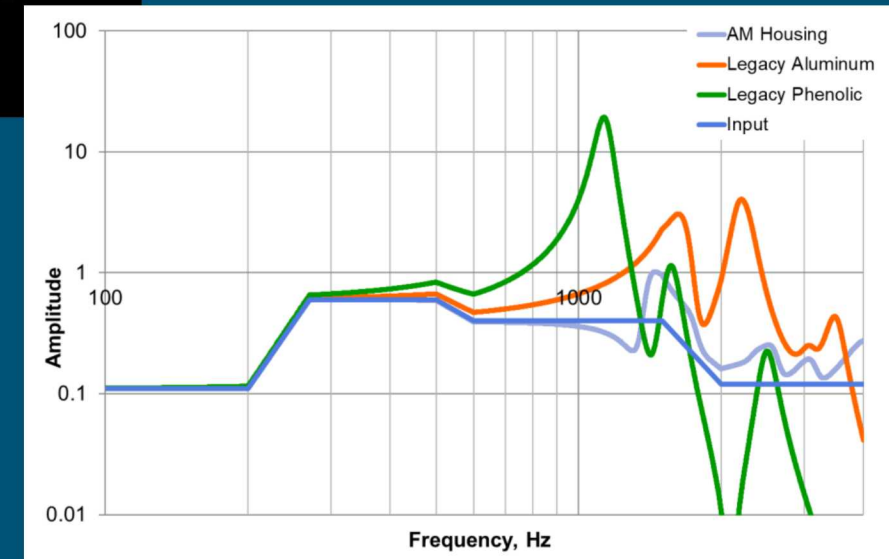
# 7 Topology Optimization

Computational synthesis for optimal material use

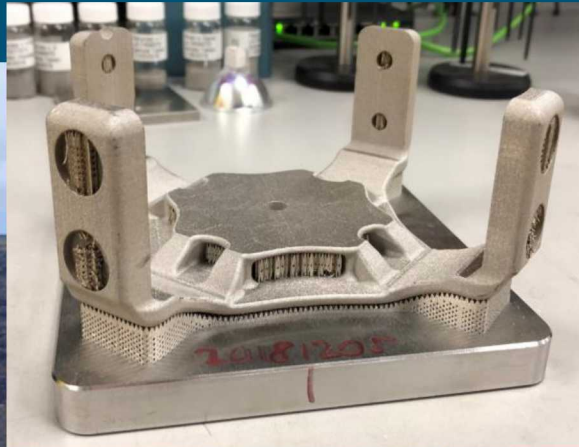
- design & optimize for performance, not mfg
- design concurrent w/simulation



*TO housing significantly reduces vibration amplitudes*



Sandia HOT SHOT sounding rocket launch from Kauai Test Facility, Hawaii



316L SS HOT SHOT top cap on plate





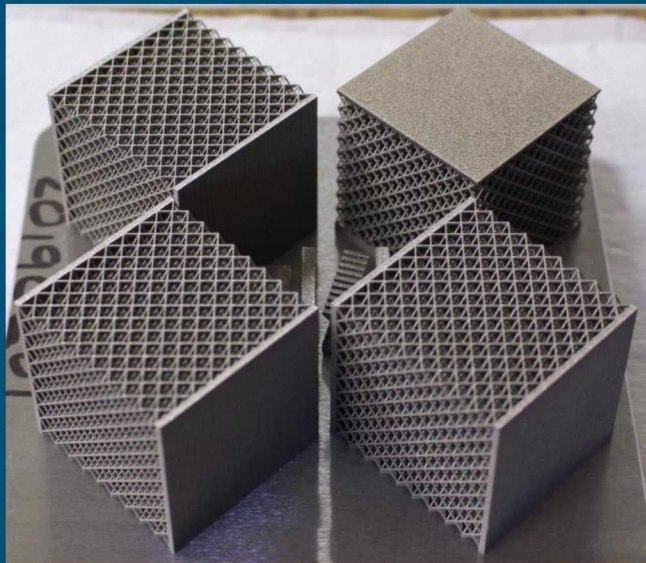
# 8 Hierarchical Materials

## Compelling design space

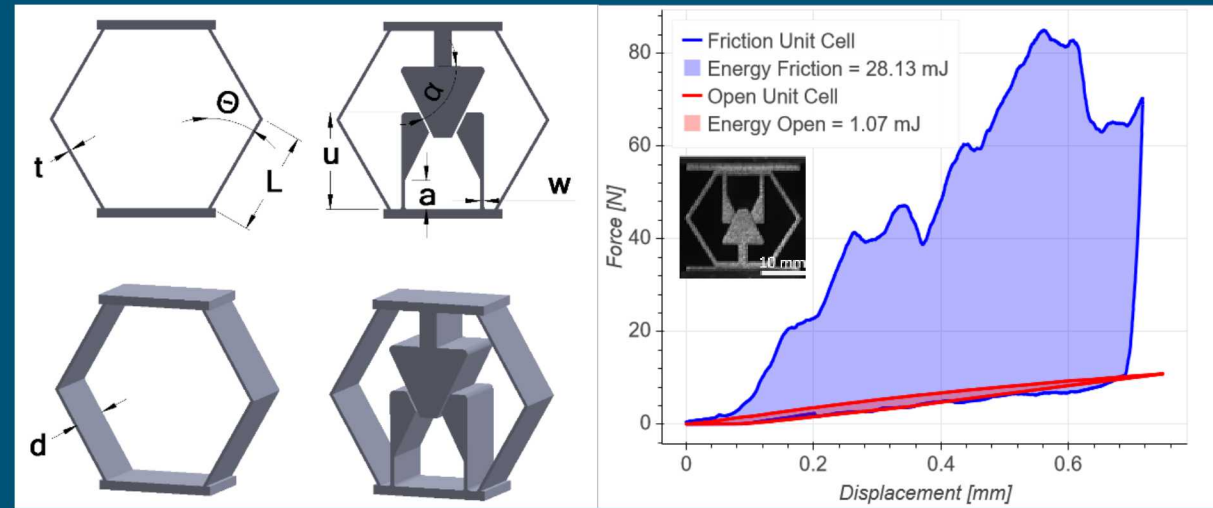
- enables performance regimes inaccessible in homogenous materials
- blurs material & structure boundaries

## Current efforts focused on dynamic impacts

- simulation, optimization, inspection & qualification



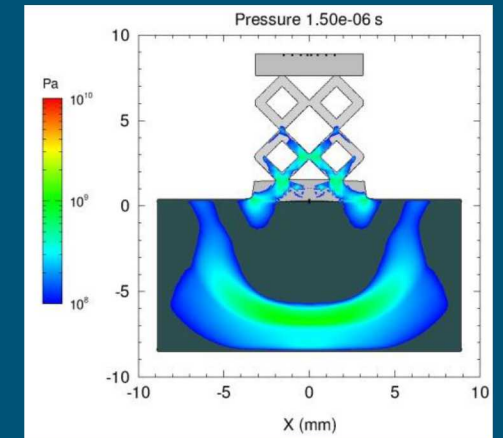
9x9x9 octet samples



Garland,  
in process



316L SS test structure  
for LANL gas gun shot



CTH simulation of 250m/s  
lattice impact



dynamic impact,  
16.3% dense BCC  
lattice

## 9 But, Complexity Isn't Free

Features tied to requirements incur costs

AM material formation concurrent w/geometry

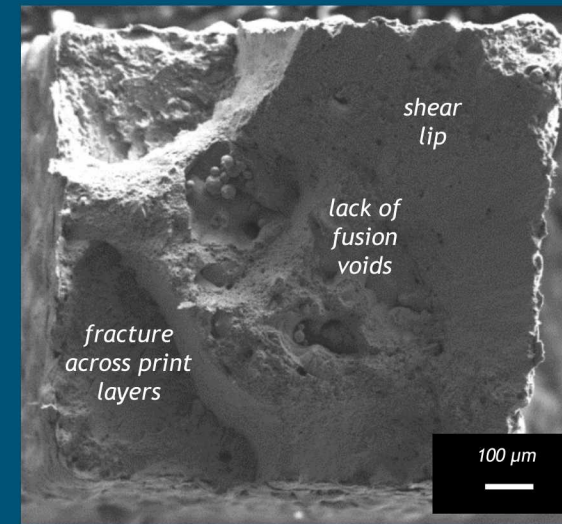
- want to predict part/material performance
- **how to ID a bad part?**
  - significant design margins and/or rigorous post-process inspection / validation

Understand mechanistic impacts on properties

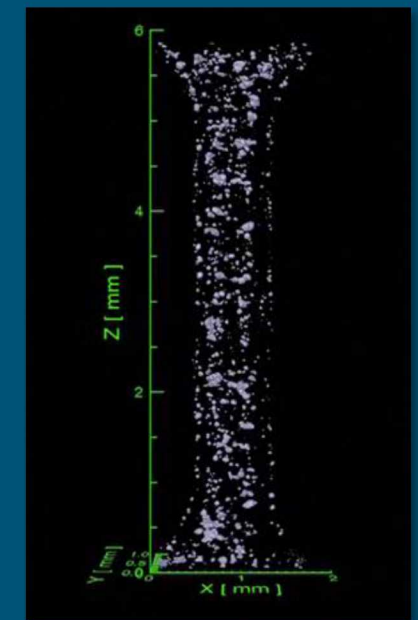
- build process-structure-property relationships to predict **margins & reliability**
- characterize stochastic response to design for **uncertainties**
- provide scientific basis for qualification of AM metals for high consequence applications

Quantify **critical** material defects & **useful** signatures

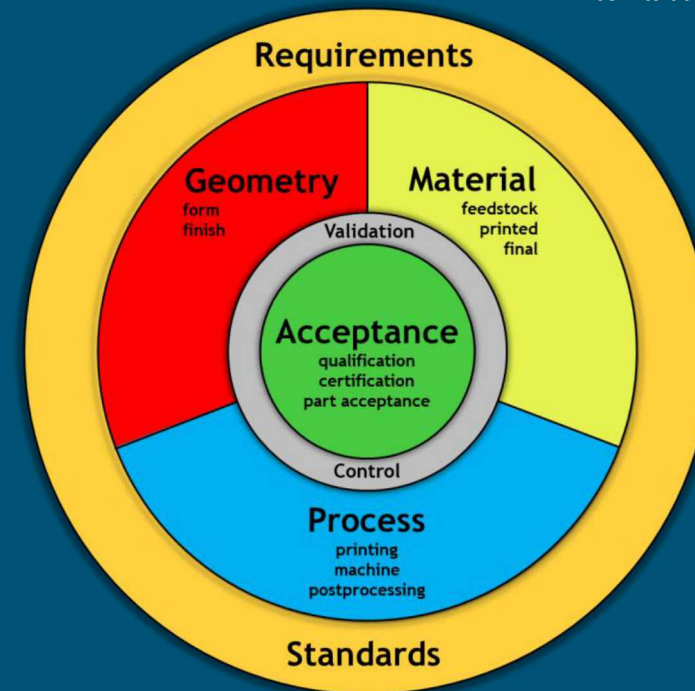
- D-tests, NDE, process monitoring, mod-sim



17-4PH dogbone fracture surface



17-4PH dogbone porosity



*elements of  
qualification  
Jared, Sci Tech and  
Appl of Metals in  
AM, 2019*



# Lack of Process Control Produced Material Uncertainty in Early AM Metals



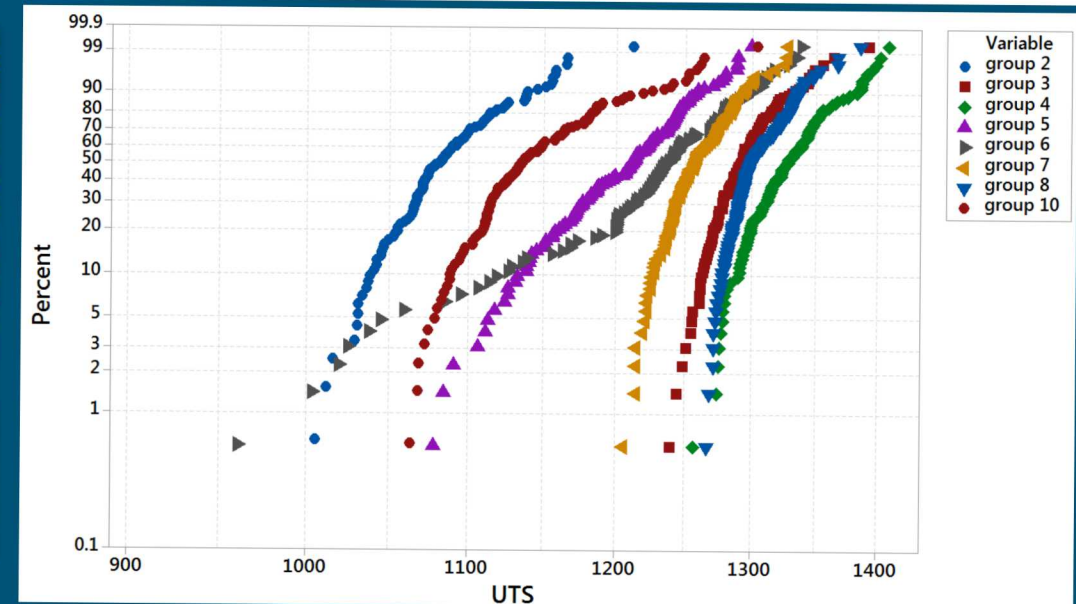
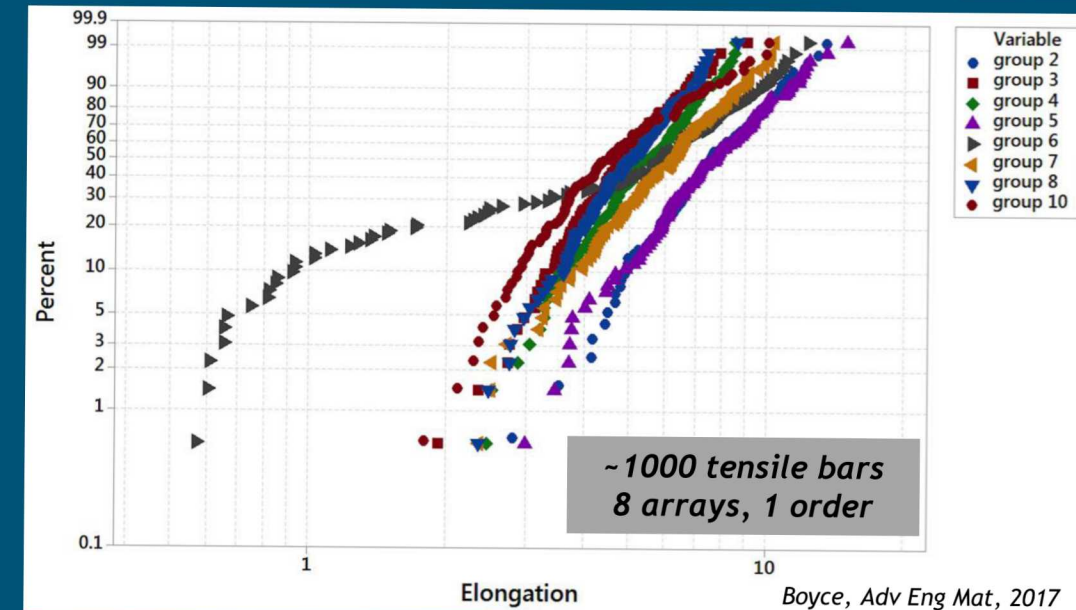
17-4PH parts from external vendor

- analysis confirmed 17-4PH composition, but unexpected microstructure

Sandia did not know

- feedstock pedigree, machine, build environment, process inputs or post-processing steps

Not-surprisingly, extensive material variability observed



17-4PH

entrapped  
gas

lack of  
fusion

20  $\mu$ m

Blue = Austenite (FCC)  
Red = Martensite/Ferrite (BCC)  
Black = non-indexed

SHT + H900  
age, 43%  
austenite

phase map  
MAG: 200x HV: 20 kV WD: 17.0 mm

90  $\mu$ m



## Controlling & logging part, build & powder cycles

- 316L stainless steel
- feedstock pedigree, build environment, process inputs, post-processing, meta data
- print / test artifacts

## Heuristic dependent

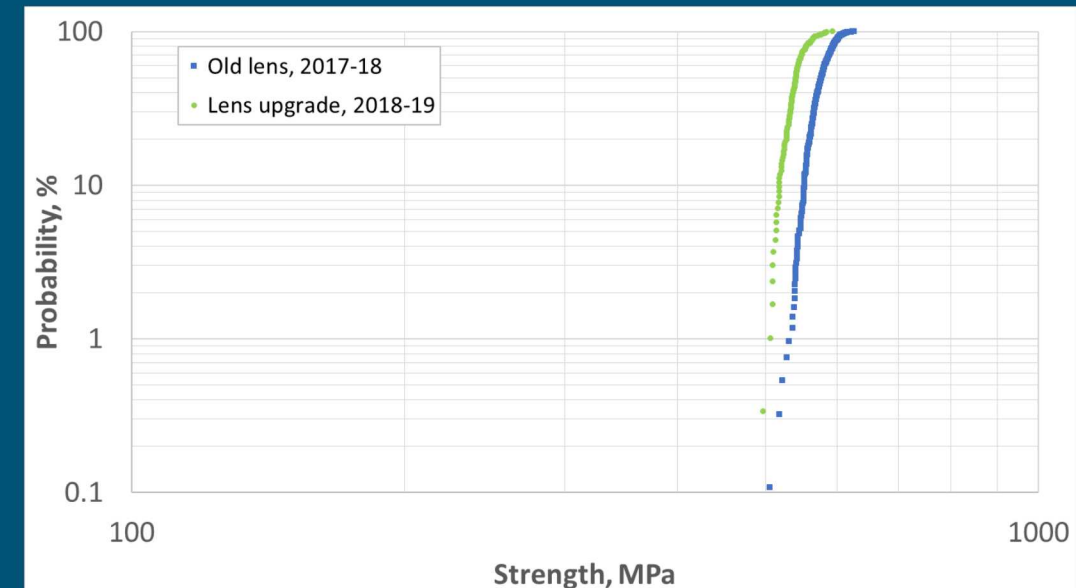
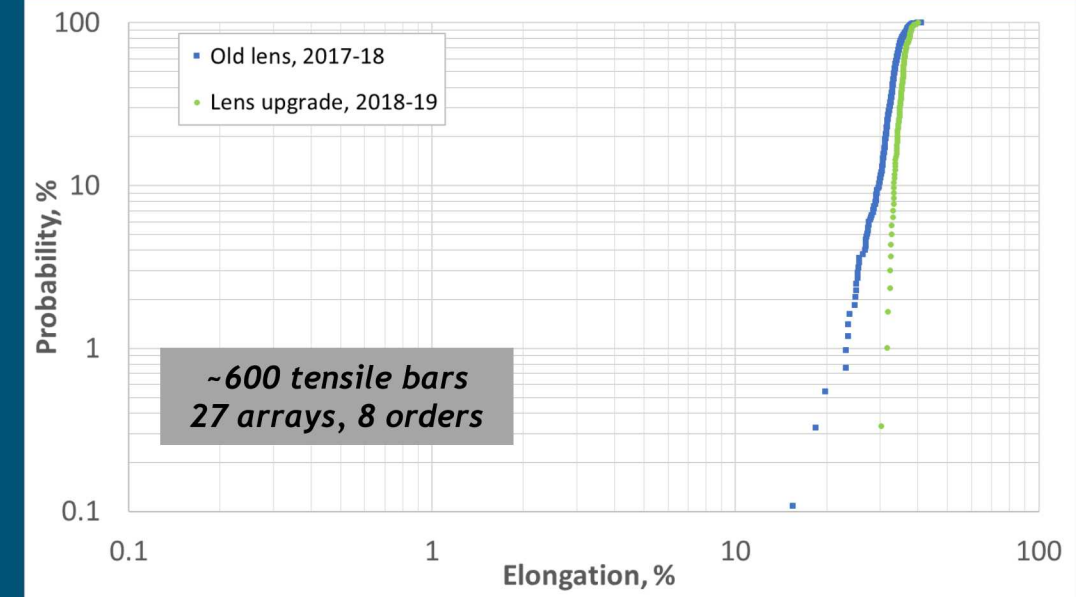
- restrictive
- time consuming
- expensive

## Desire accelerated cycles

- tolerances & uncertainties

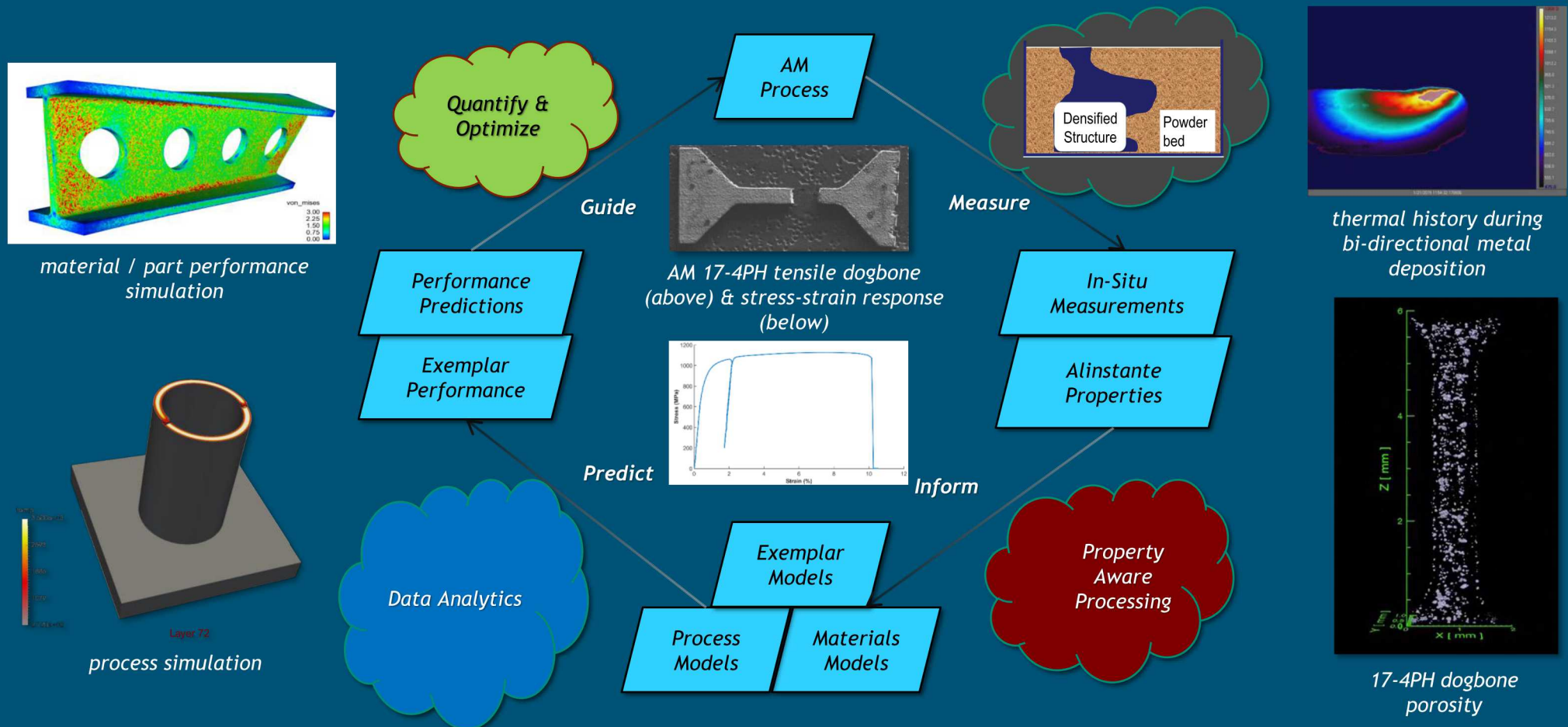


build plate w/process artifacts



## “Changing the Engineering Design & Qualification Paradigm”

- leverage AM, in-process metrology & HPC to revolutionize product realization
- accelerating design to production



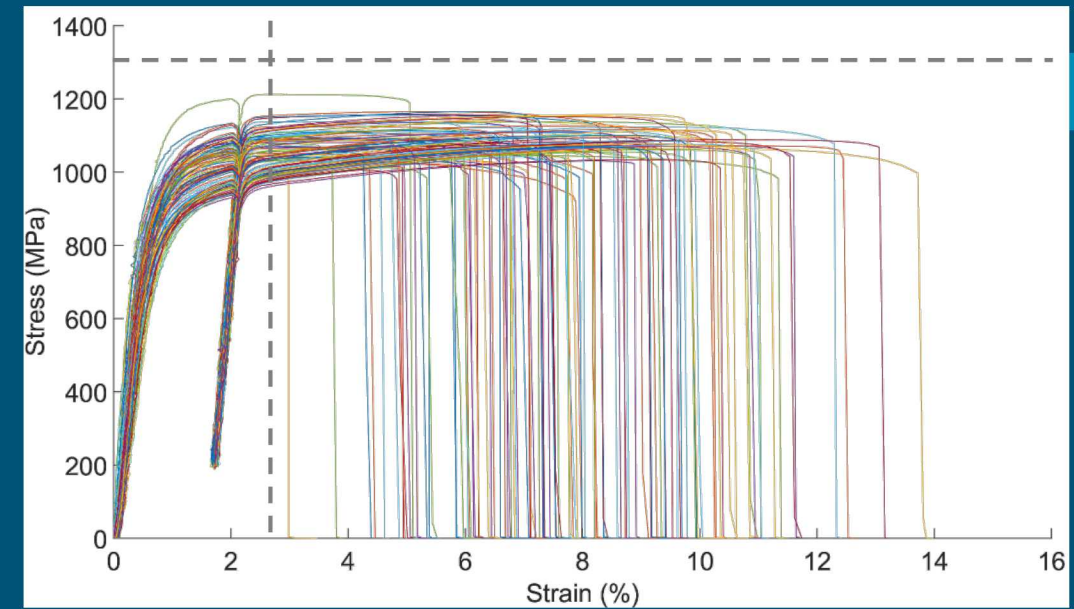


# Stochastic Response of AM Metals

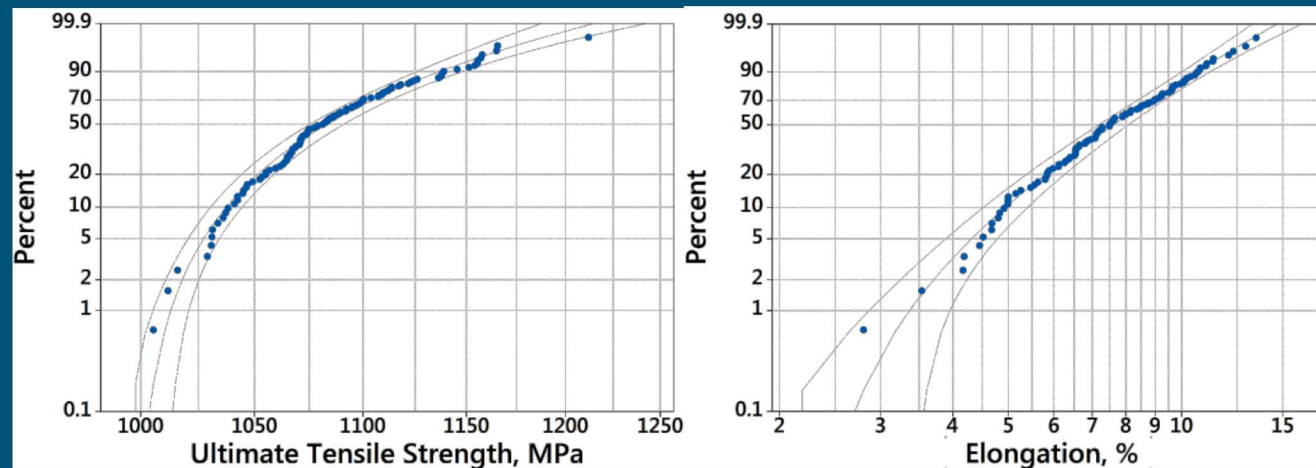
## Defect dominated failure

- 3-parameter Weibull informs design threshold
- ductile dimples & shear rupture planes
- voids & lack-of-fusion boundaries are likely crack nucleation sites

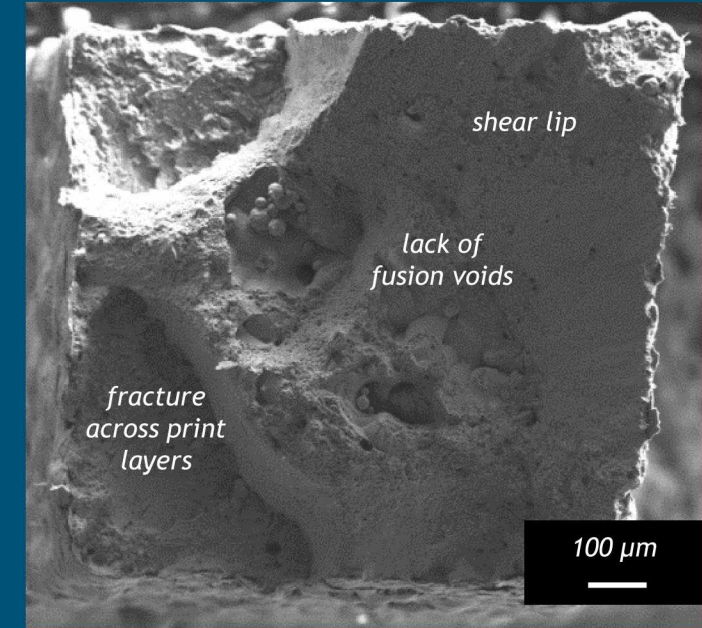
How to capture efficiently & accurately?



110 stress-strain curves for 17-4 PH after SHT+H900



material performance fit to 3-parameter Weibull distributions



failure at 2% elongation, SHT+H900



# High Throughput Tensile Testing (Gen I)

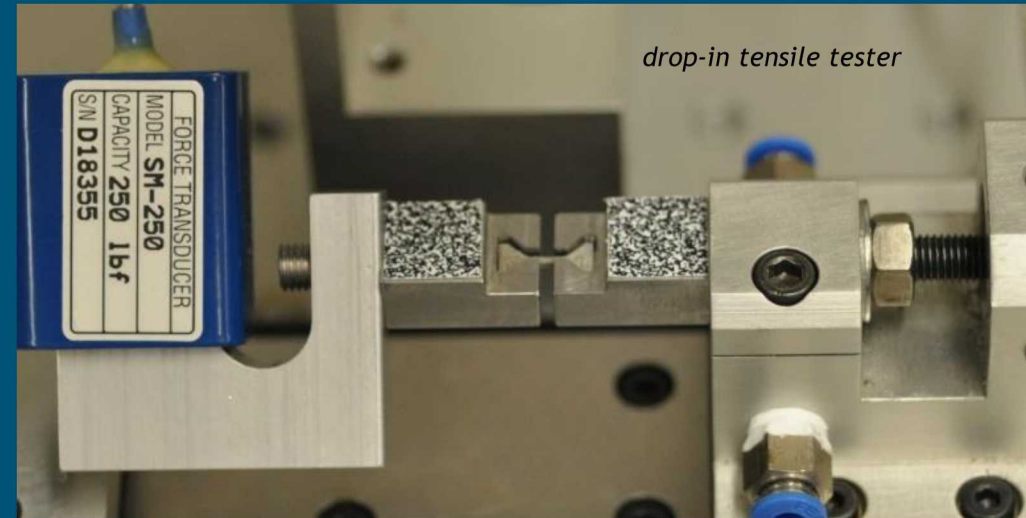
Monolithic build w/110 dogbones

- custom design per ASTM

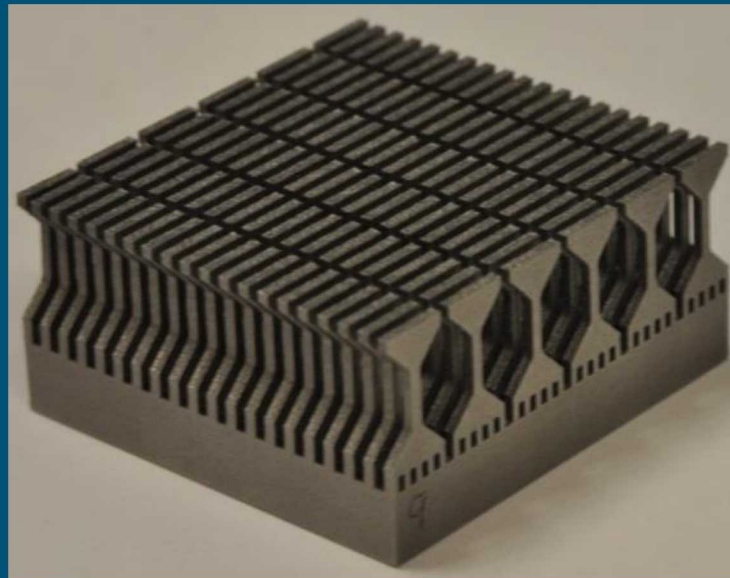
Digital image correlation (DIC)

Necessary to rapidly capture material distributions

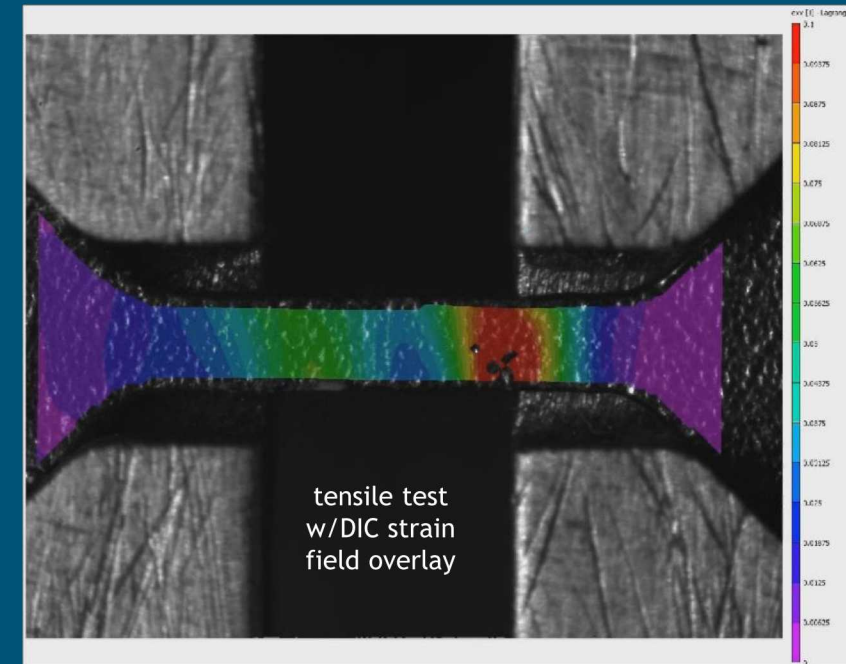
- applicable for the lab & production



Salzbrenner, Journal of Materials Processing Technology, 2017



high throughput test sample w/120 dogbones,  
1x1mm gage x-section



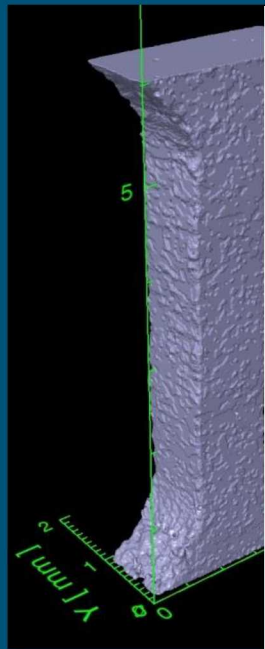
# Tying Structure (Porosity) to Properties

Extensive work using computed tomography (CT)

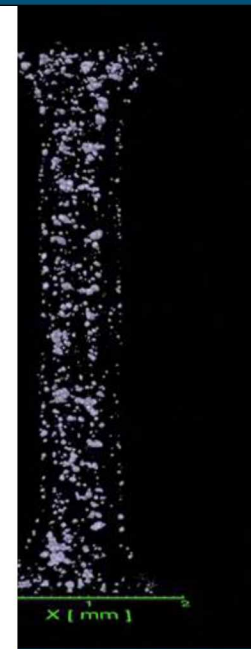
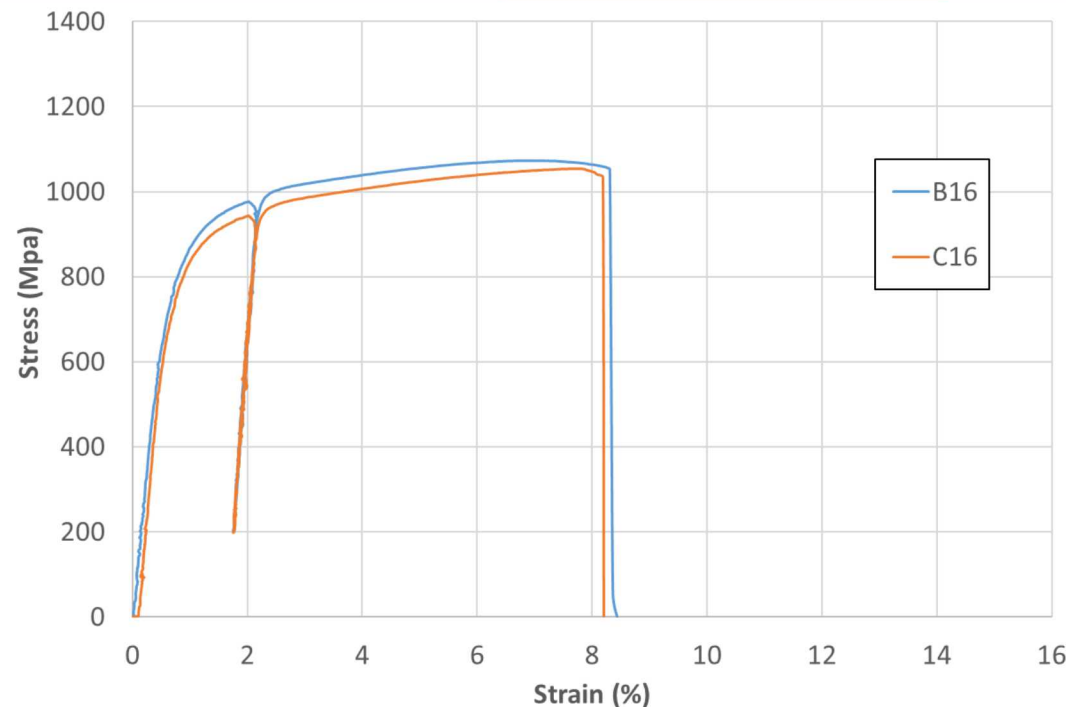
- multiple potential metrics exist
- correlations are immature

Prediction of uniformly dispersed porosity fields is difficult

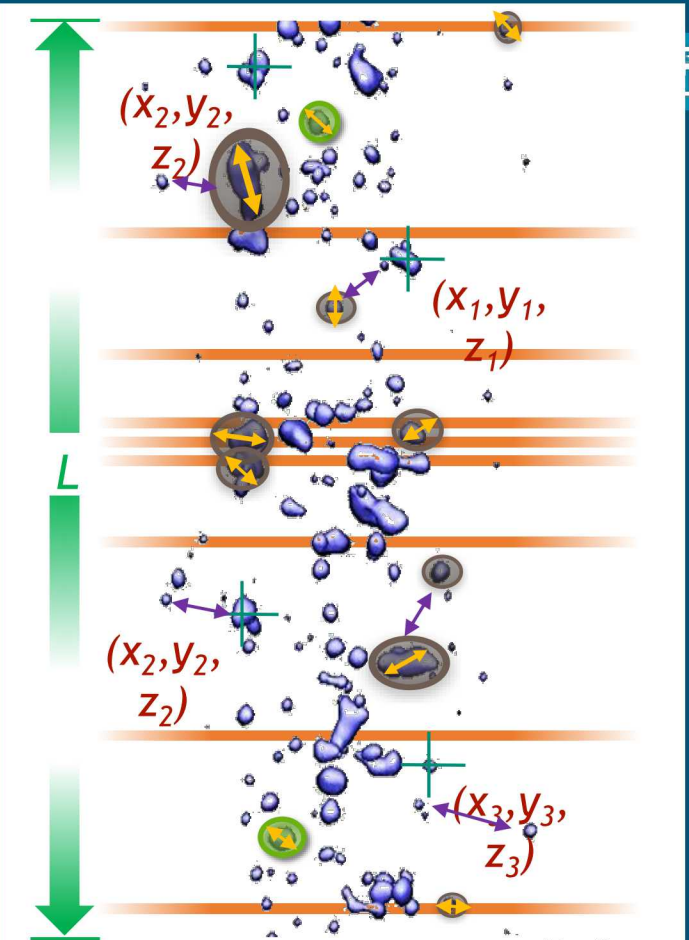
- behavior falls within an expected performance distribution



17-4PH dogbone B1



porosity map



Madison,  
QNDE, 2018

- Total volume of defects ( $V_{tot}$ )
- Pore volume fraction ( $V_{fract}$ )
- Spatial location of pores ( $x, y, z$ )
- Total number of defects ( $N$ )
- Total defects/length ( $N/L$ )
- Average defect volume ( $V_{avg.}$ )\*
- Average equivalent spherical diameter ( $ESD_{avg.}$ )\*
- Average cross-sectional area ( $CSA_{avg.}$ )\*
- Average nearest neighbor distance ( $NND_{avg.}$ )\*



# Gross Defects Drive Performance Outliers

Failure initiates near large pores

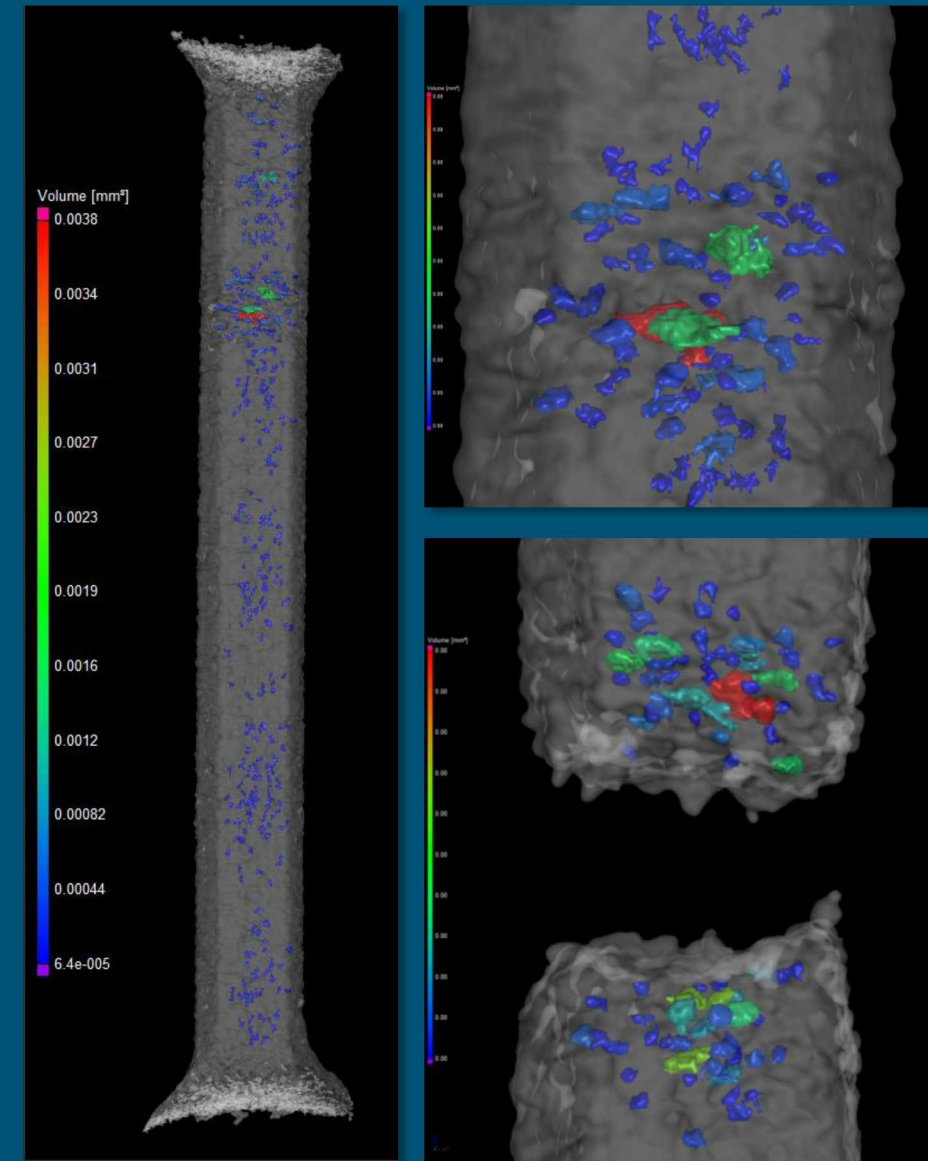
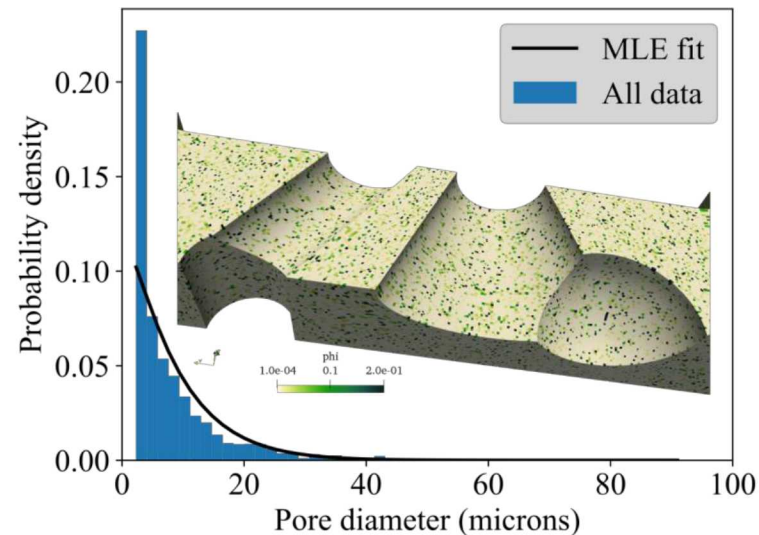
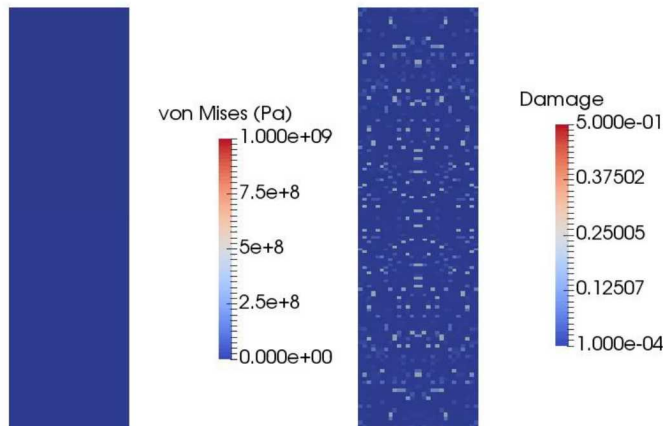
Potential tolerance bound

- design requirements
- CT inspection
- process monitoring

Performance simulation represent power analysis & diagnostics tools

*representing porosity as initial damage*

*Johnson, Int J Fract, 2019*



*1x10mm 316L SS tensile dogbone (left) from digital volume correlation (DVC), gross porosity region (top right), failure region (bottom right)*



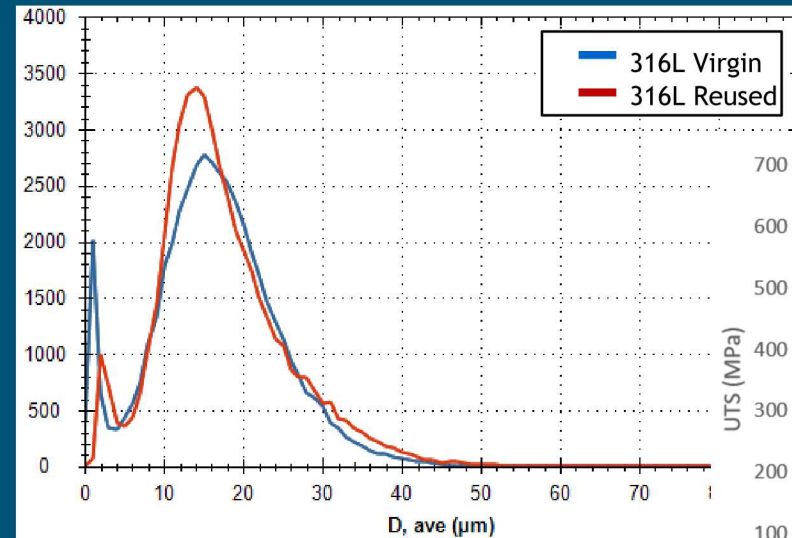
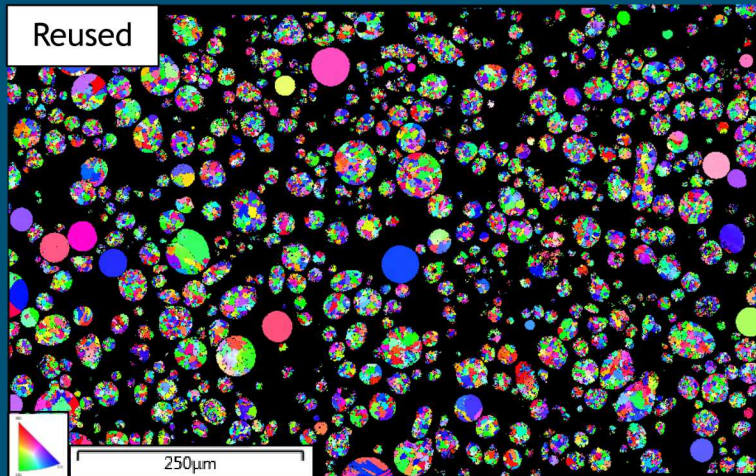
## 17 Monitoring Powder Reuse

Tracking powder size, morphology & EDS composition w/reuse

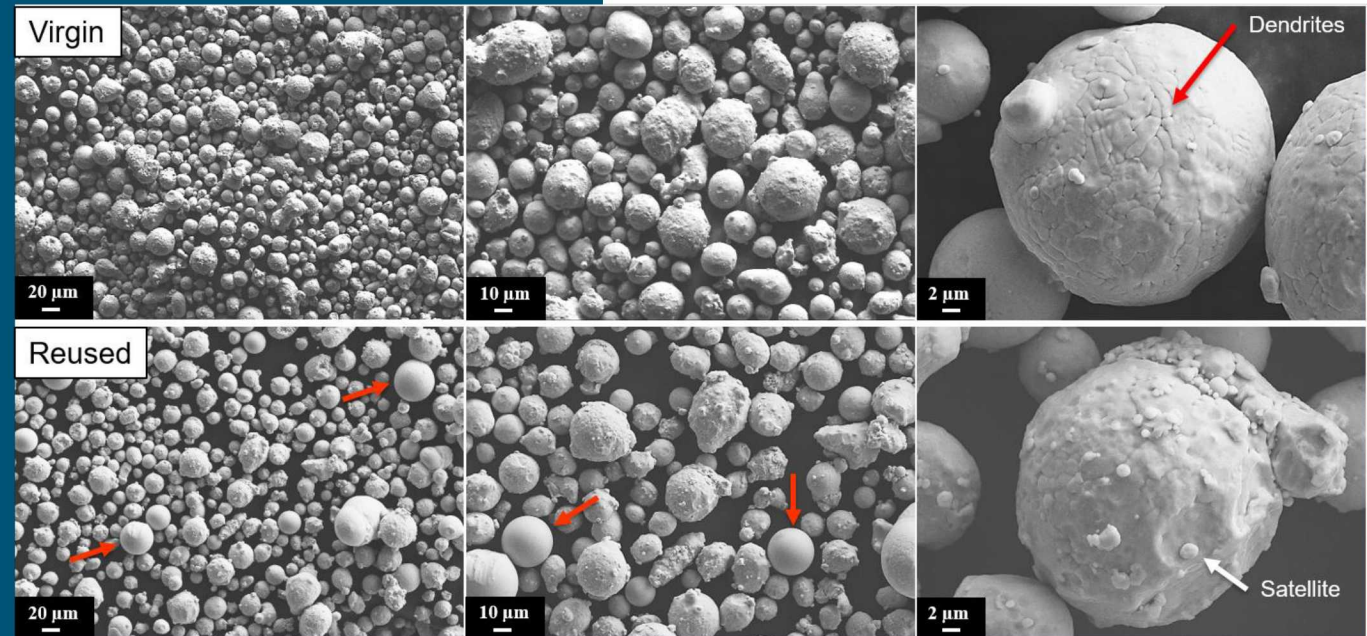
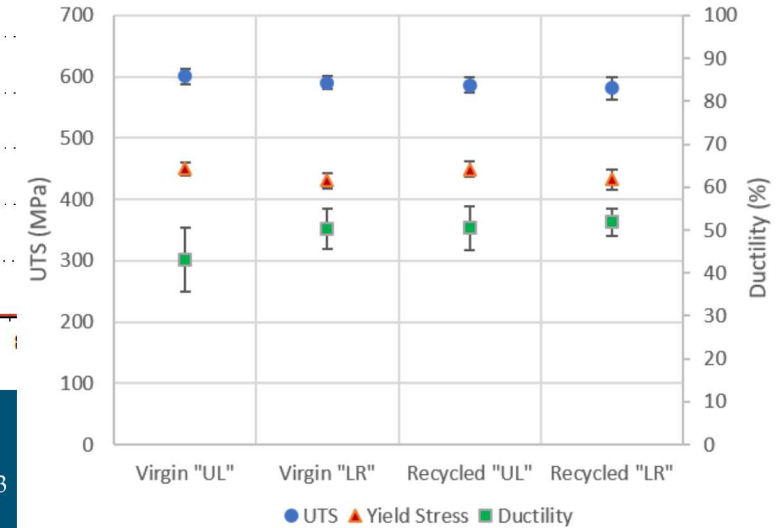
- satellites & agglomerates increase
- observe highly spherical, ferrite particles
- increase in fines & reduction in larger particles
- collected over 30 reuses w/powder under Ar

Material properties remain stable

316L SS is a robust material for processing & properties

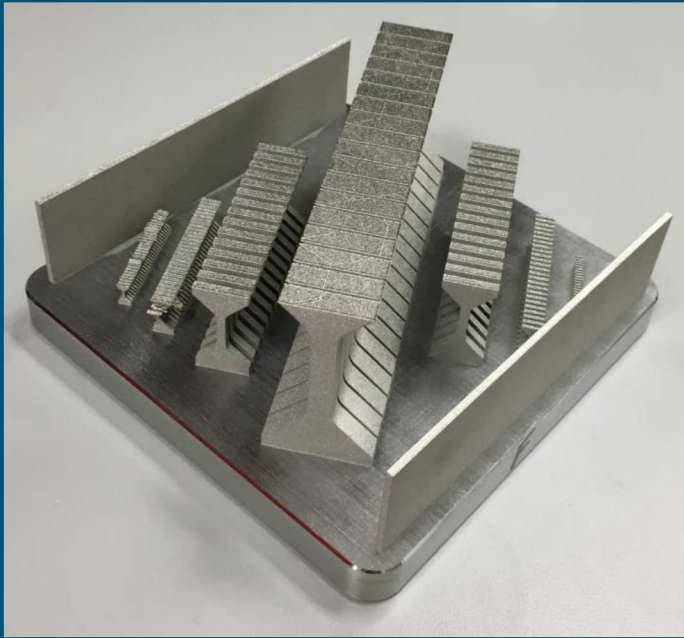


Virgin: 7.90 g/cm<sup>3</sup>  
Reused: 7.81 g/cm<sup>3</sup>

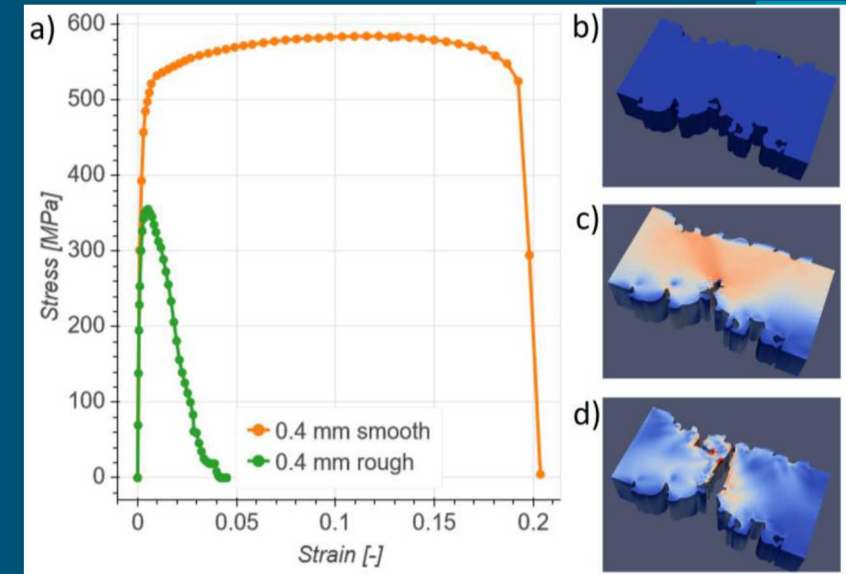
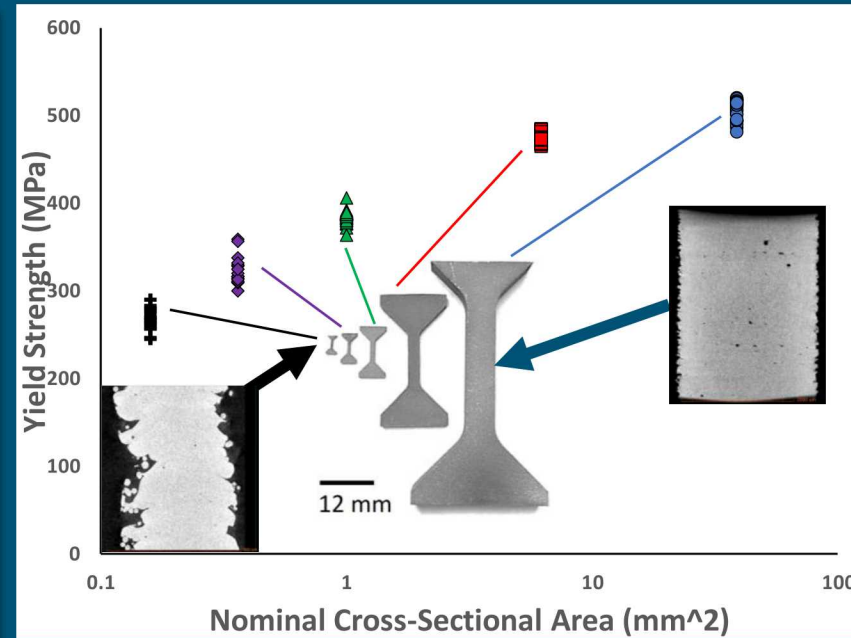




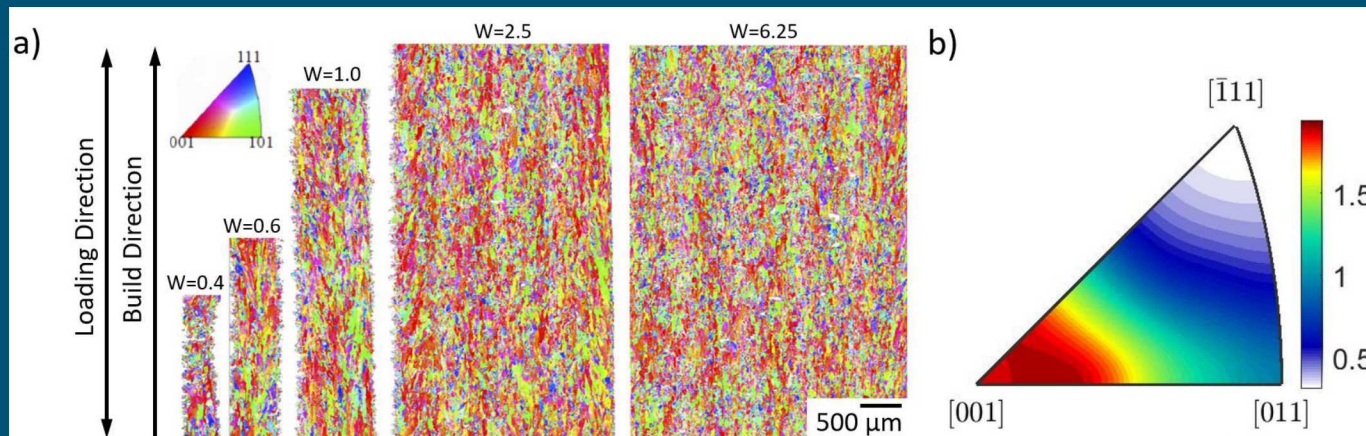
## Size Effects



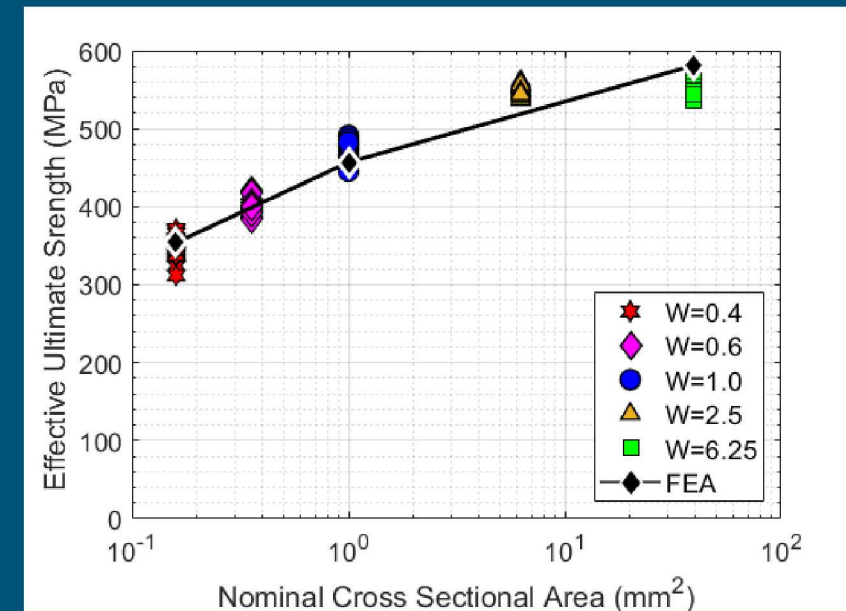
Prox 200 build plate w/tensile samples



surface roughness dramatically reduces strength of features with similar size scale



microstructure is invariant w/sample size



FEA model incorporating surface finish correlates well to experiments



# Process-Structure-Property Process Maps

## Parameters

- 316L stainless steel
- laser power: 10-240W
- velocity: 50-2800mm/sec
- layer thickness: 30, 40 $\mu$ m
- average powder diameter: 15, 25 $\mu$ m
- laser focus offset: -1.5mm below focus to +3.5mm above

## Experiment forms

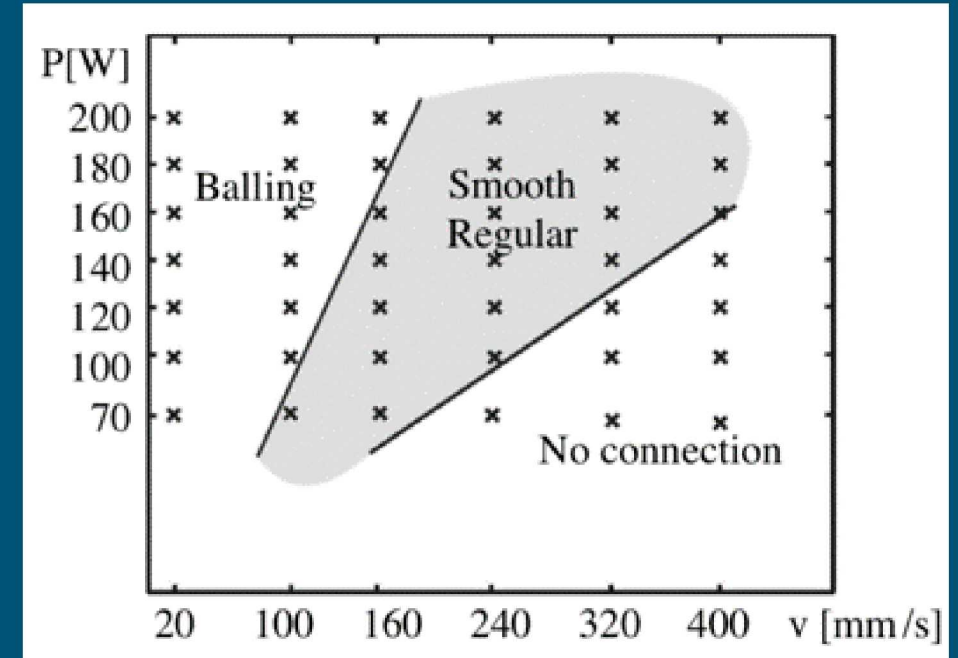
- line scans, area pads, density cubes, HTT tensile, Charpy

## Performance metrics

- surface finish, form error, density, tensile properties, Charpy toughness, microstructure
- where are optimal process settings?

## *The Influence of Process Variables on Physical and Mechanical Properties in Laser Powder Bed Manufacturing*

- Josh Koepke, MS Thesis, UNM Dept of ME, 2019



Kruth, *Rapid Prototyping Journal*, 2008

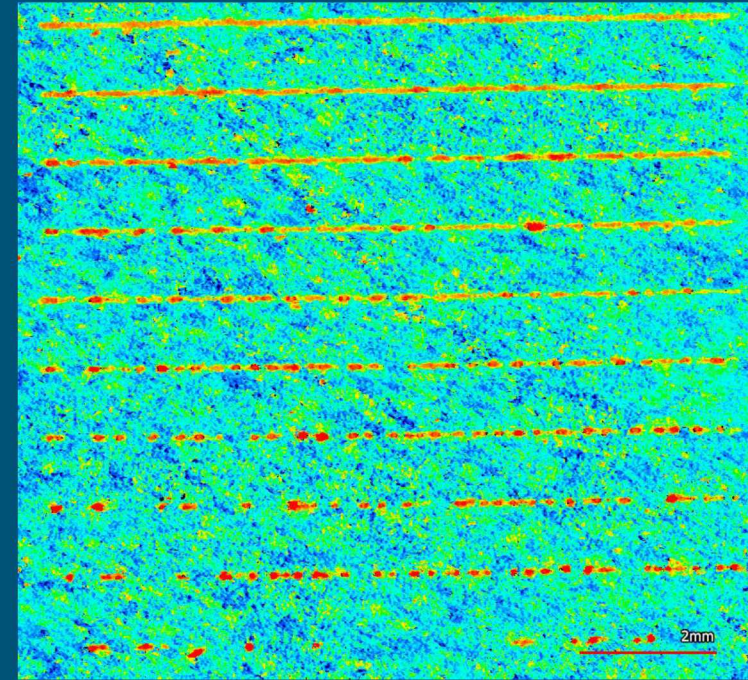
## Line Scans

Substrates: bare plate, powder layer on 20 layer AM pad

- 60 lines on each substrate, 1.0 cm long
- power: 25-175W
- velocity: 250-2500mm/sec

Simplistic first step, but quick & informative

- capturing melt pool geometry via metallography
- useful to define nominal process boundaries for any material
- used to establish relevance of simple Rosenthal model



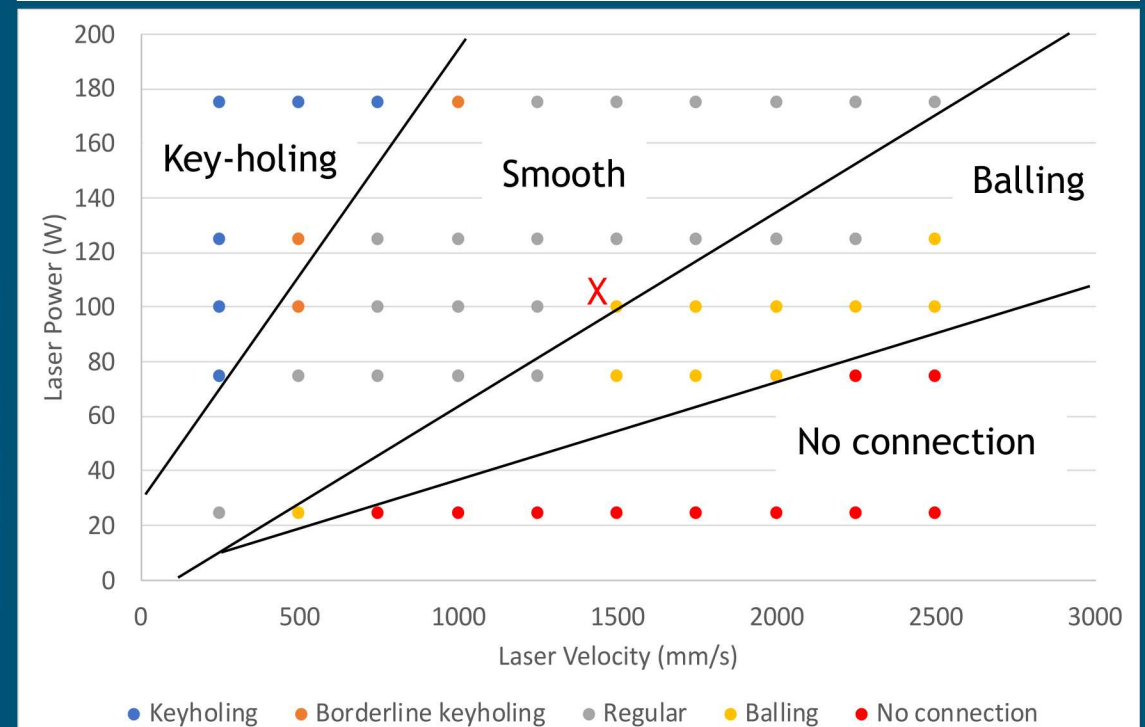
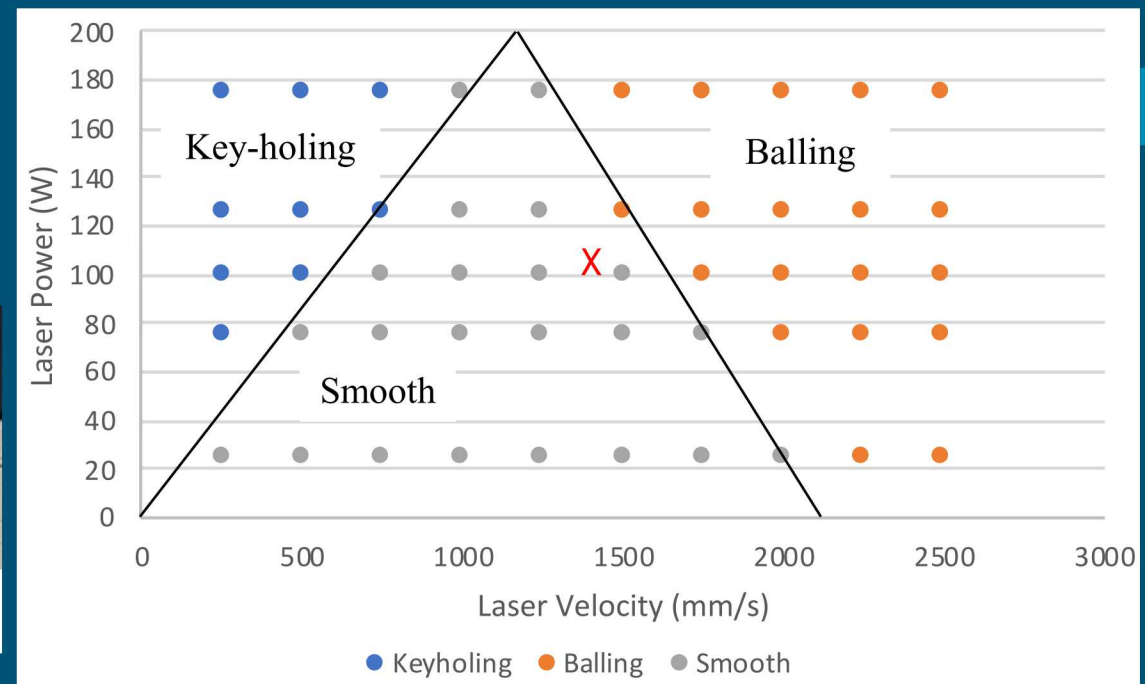
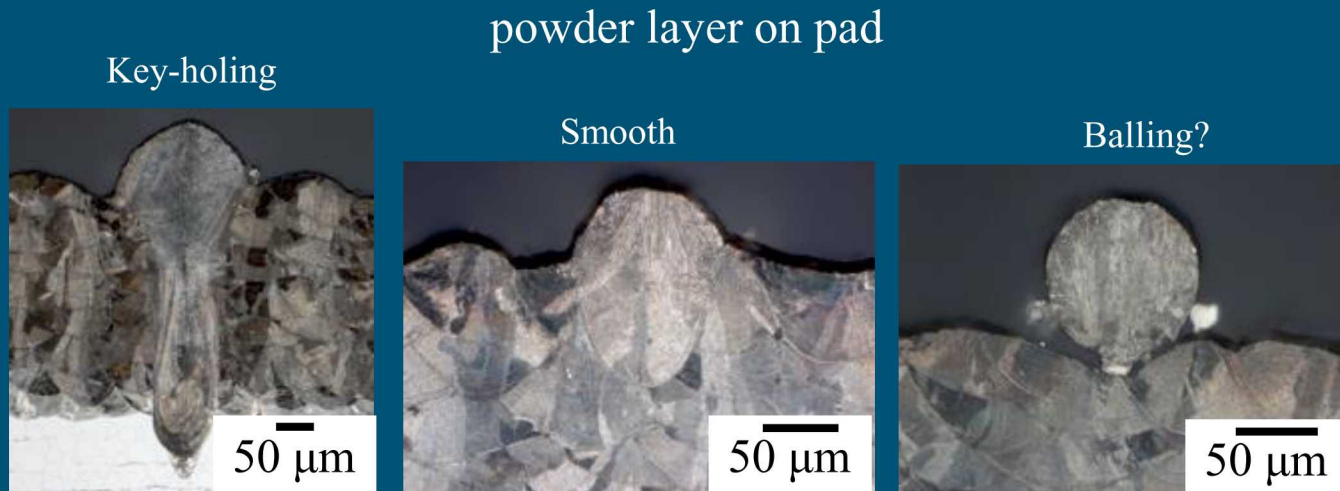
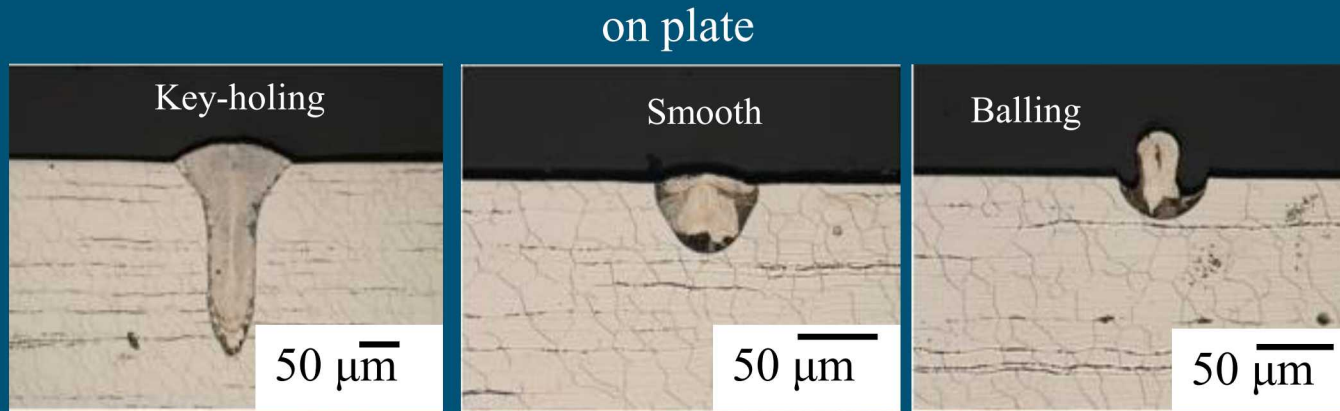
100 W, 250 to 2500 mm/s on pad



melt pool on plate, 100W, 1500 mm/s



# Line Scans: Plate vs. Powder Layer on Pad



Predicts heat distribution from a moving heat source during welding

- Provides an estimated melt geometry

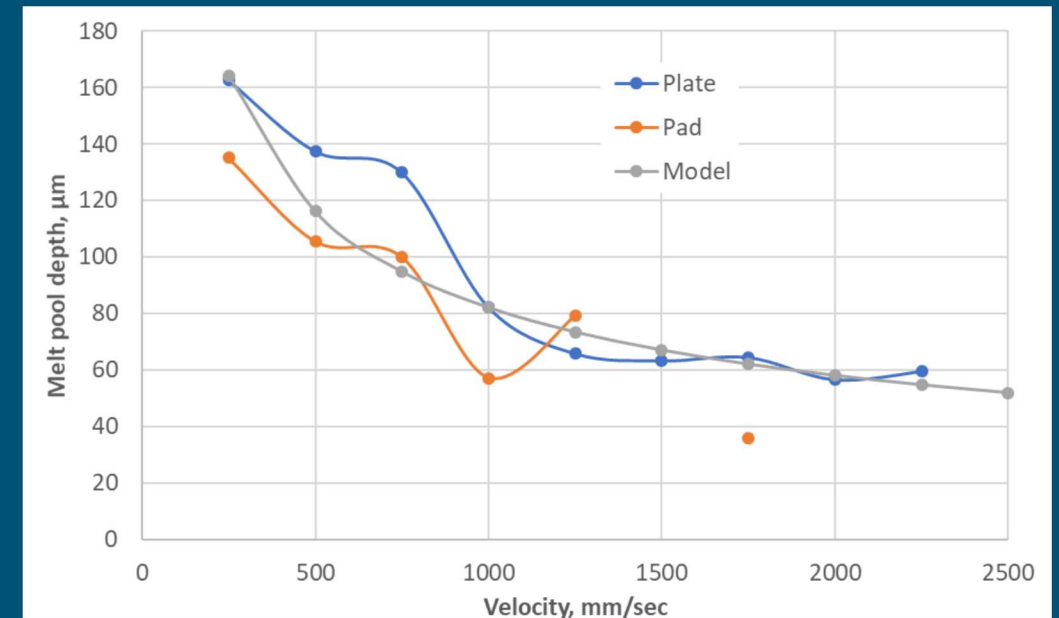
$$T_f - T_0 = \frac{q}{2\pi k R} e^{-\frac{v(w+R)}{2\alpha}}$$

- Modified (“Prediction of lack-of-fusion porosity for powder bed fusion” Tang et al)

$$W = 2 \sqrt{\frac{2q}{e\pi\rho C_p(T_f - T_0)v}}$$

- W – width of melt pool
- q – laser power \* absorptivity of material
- v – laser velocity
- $T_f$  – melting temperature

Again, very simple & easy



melt pool depth at 100W



# Density Cubes

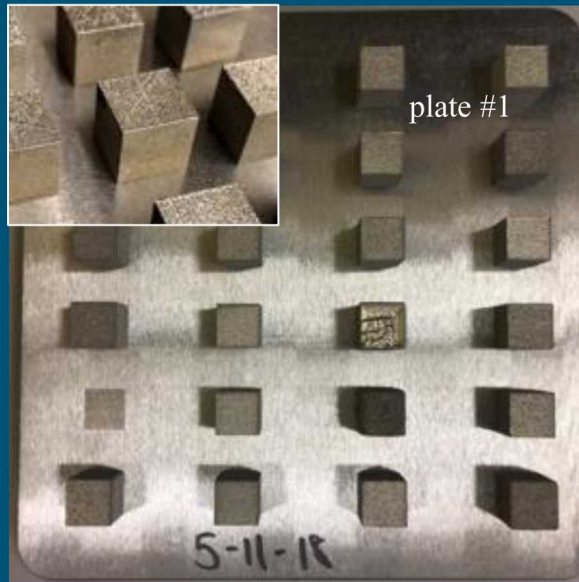


1cm cubes, 24 per plate

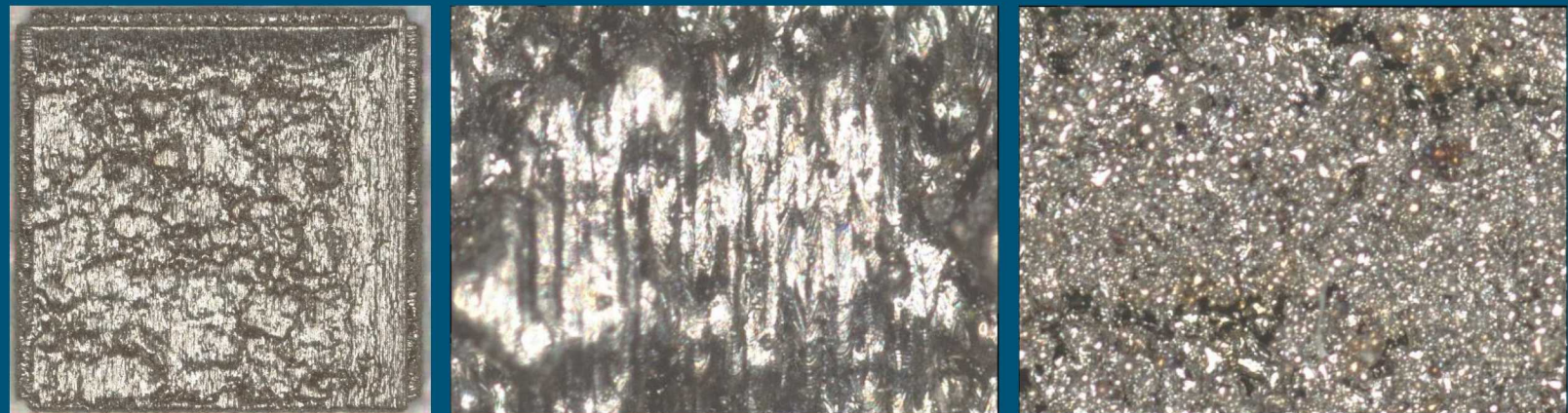
- 10 plates, power, velocity, focus offset, layer thickness, powder size, variation across plate

Bulk material measurements

- density (Archimedes)
- top & side surface form & finish
- microstructure: optical, EBSD



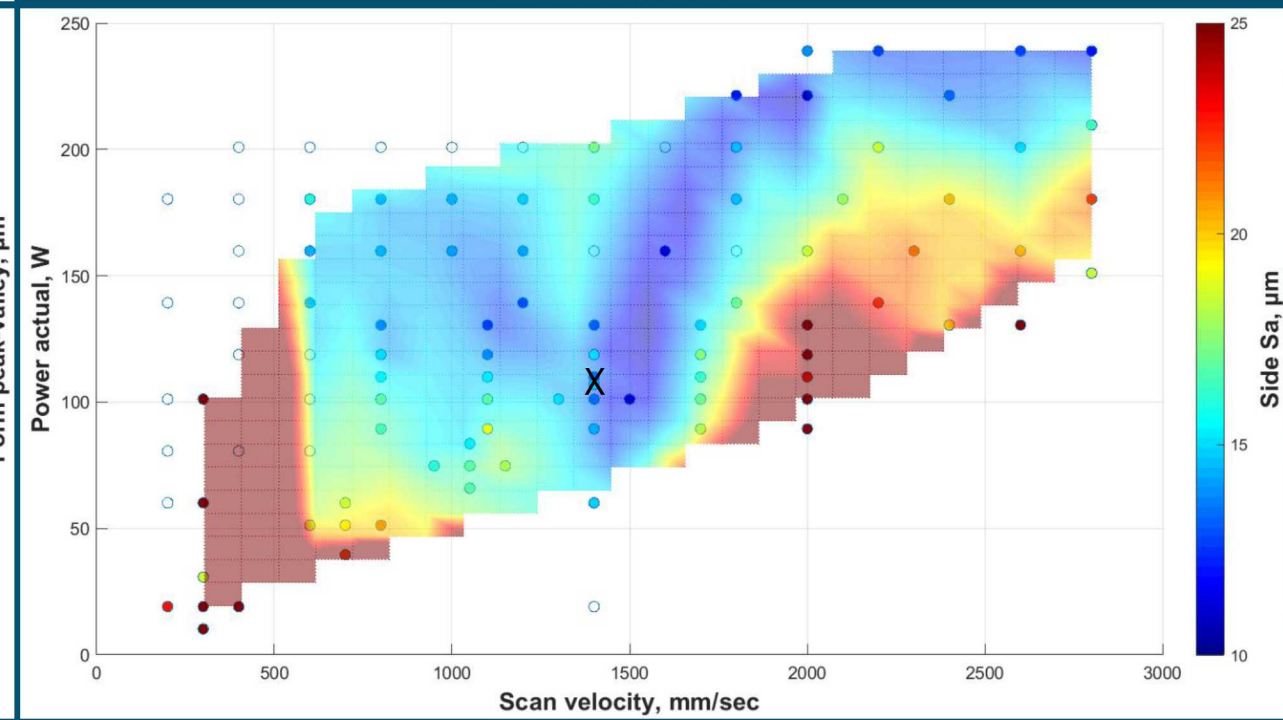
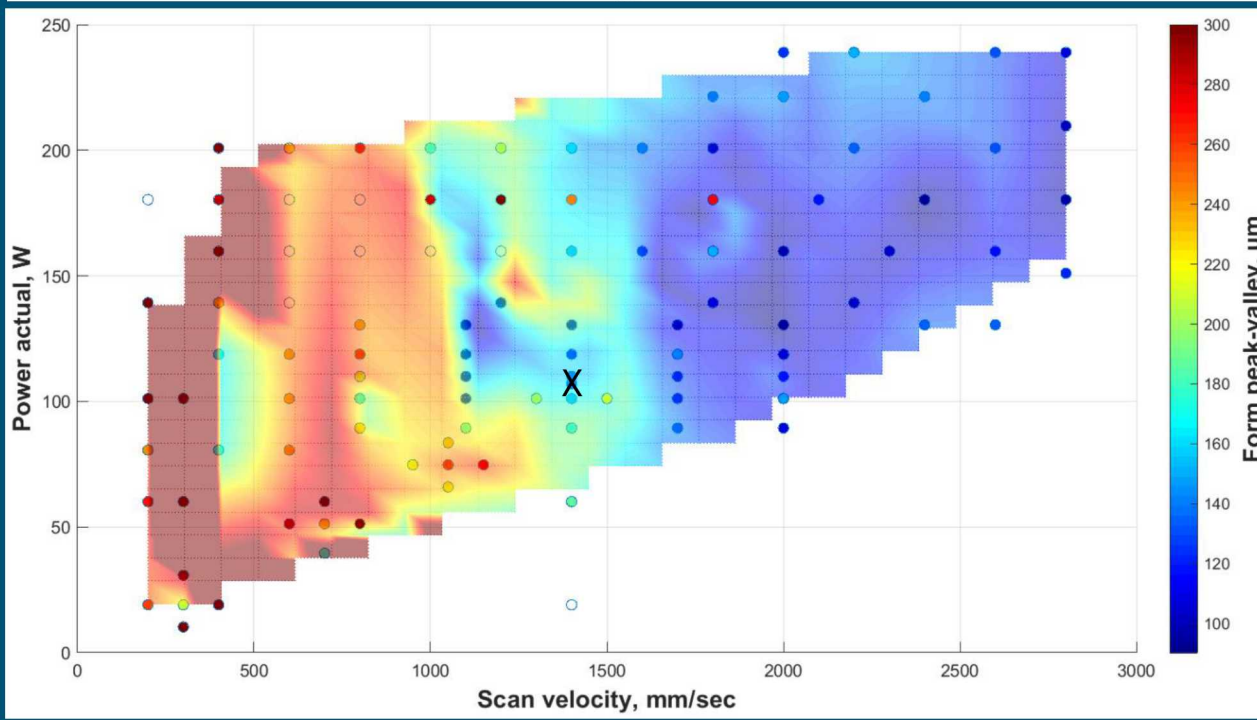
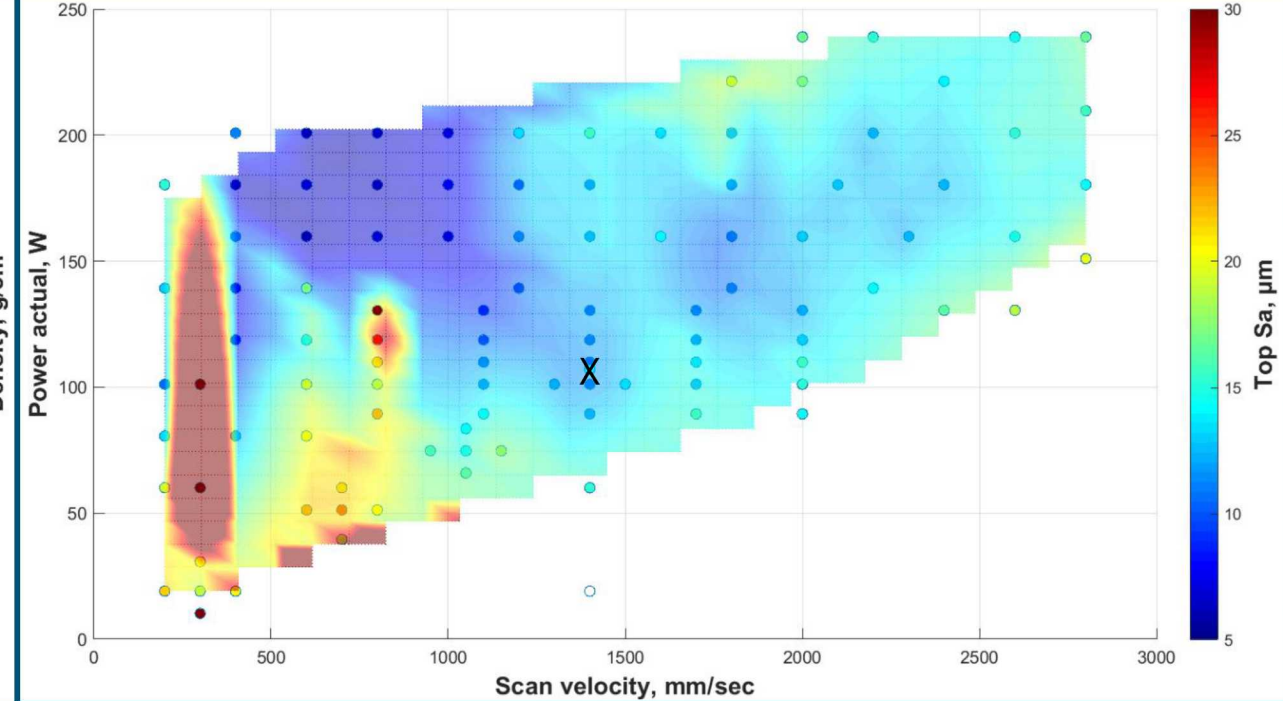
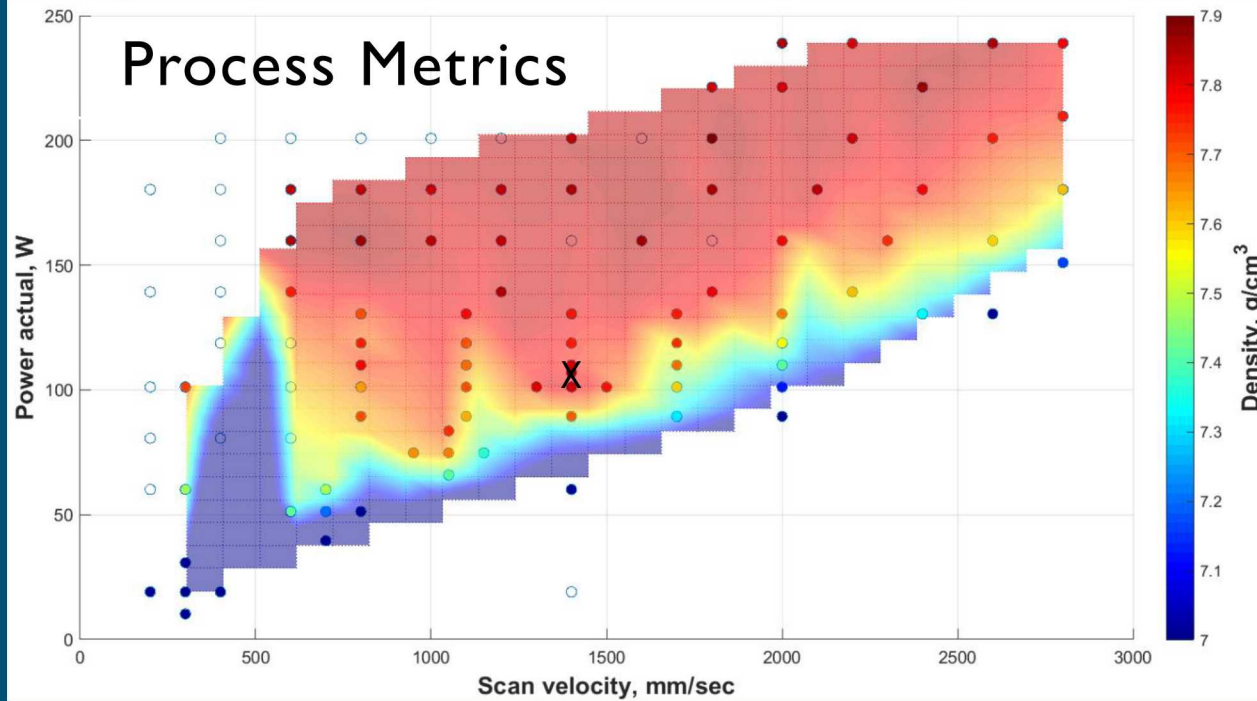
100.2W, 1500mm/sec 209 $\mu$ m PV form (left), top 13.5 $\mu$ m Sa roughness (center), side 9.98 $\mu$ m Sa roughness (right)



60.1W, 300mm/sec 891 $\mu$ m PV form (left), top 73.8 $\mu$ m Sa roughness (center), side 38.1 $\mu$ m Sa roughness (right)

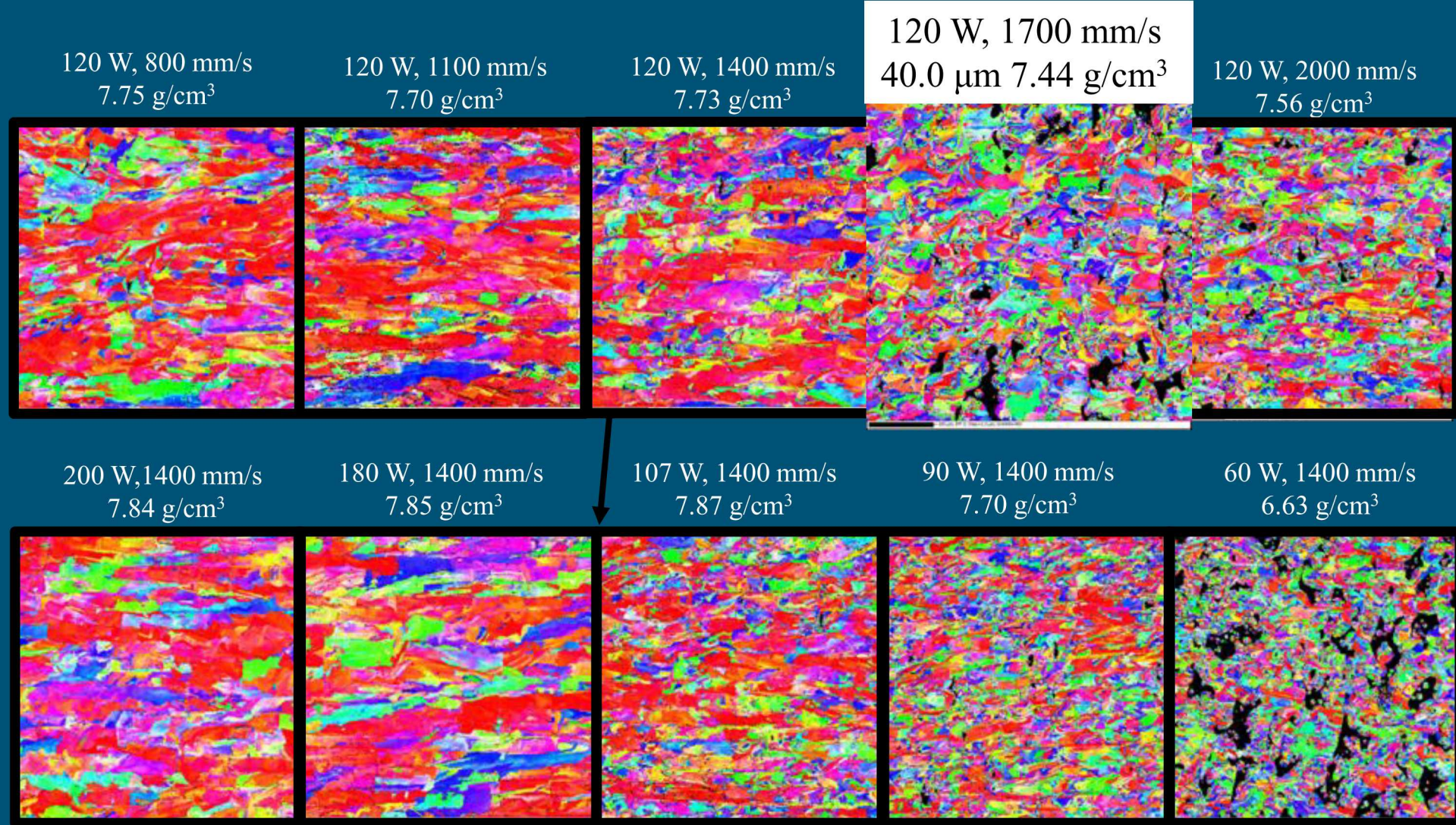


# Process Metrics





# Material Microstructures, 30 $\mu$ m Layer Thickness



1 mm x 1 mm

decreasing energy density





# High-Throughput Tensile Properties

1x1x4mm gauge section, 10 or 25 dogbones/array

- five build plates
- varied power, velocity, powder diameter
- Gen2 HTT system

>500 dogbones tested

- density (Archimedes)
- surface roughness
- mechanical properties
  - UTS, YS, modulus, ductility

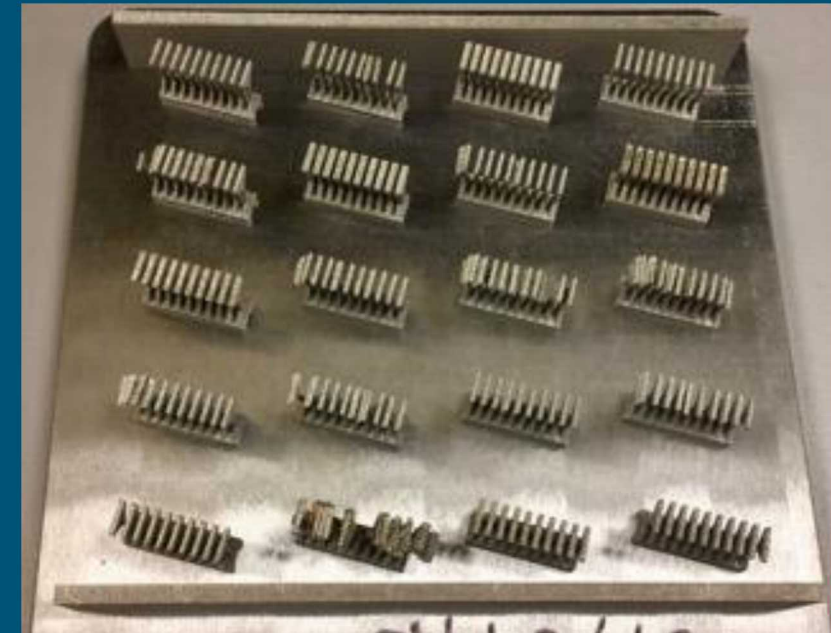


*Heckman, Mat. Sci. Eng. A, 2020*

*316L SS dogbone array with 25 dogbones*

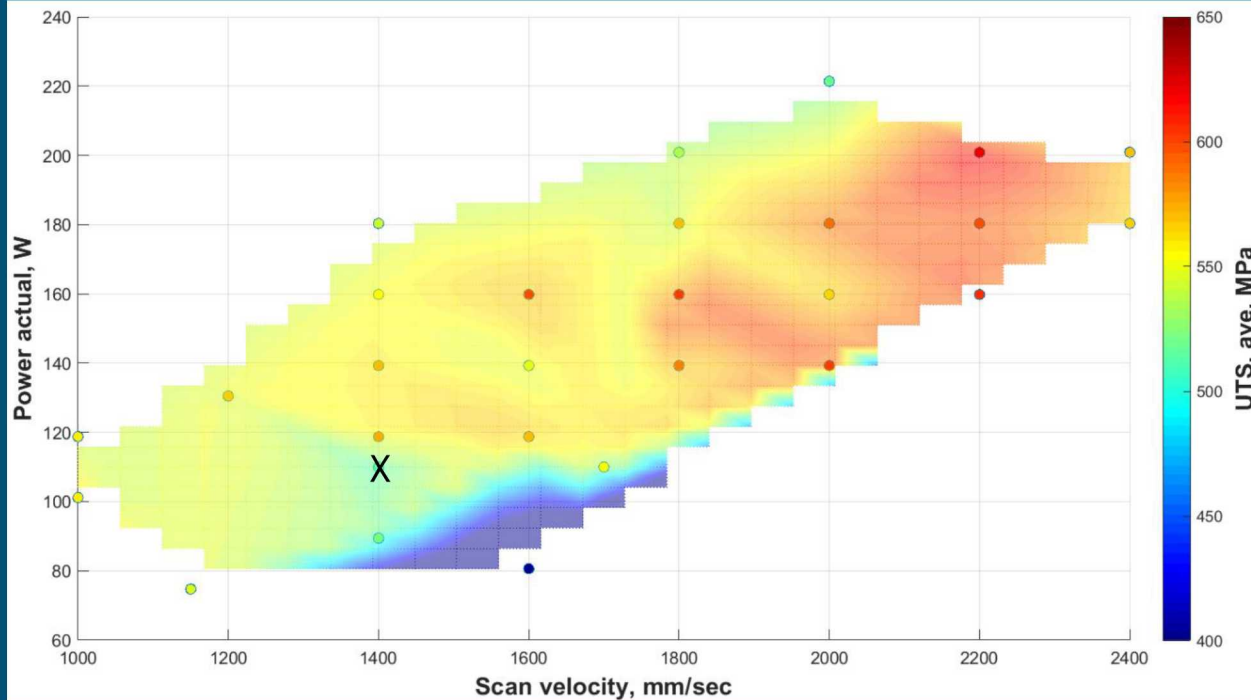
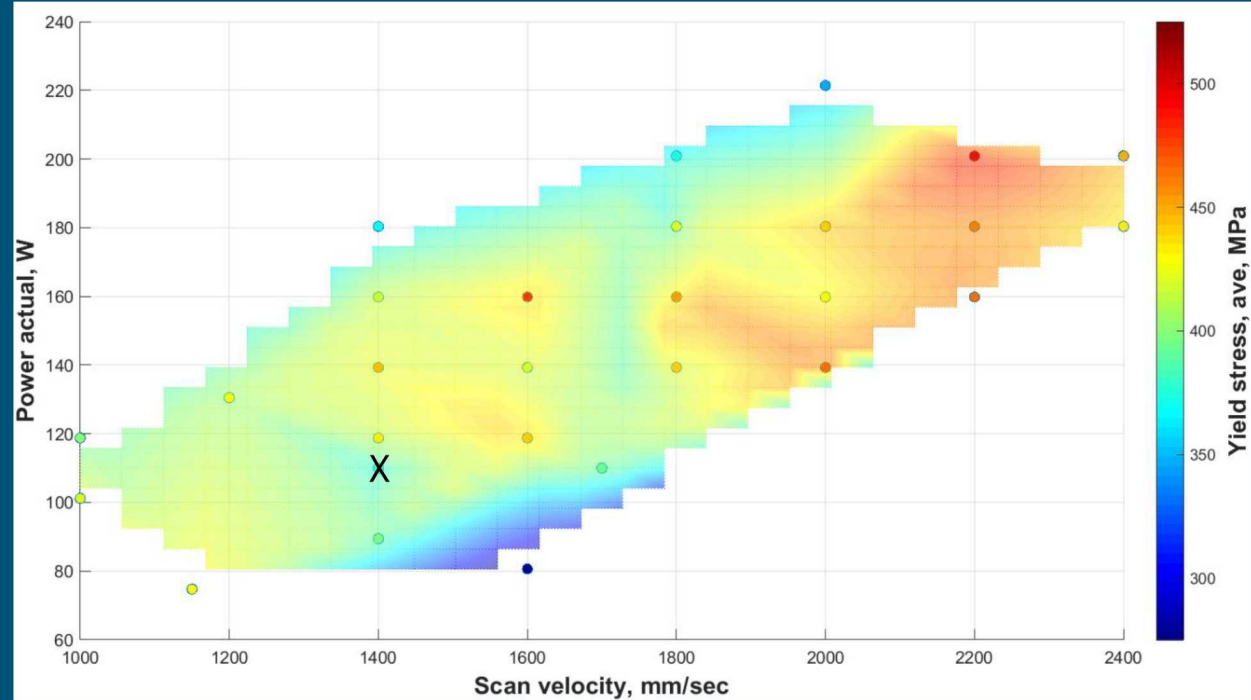
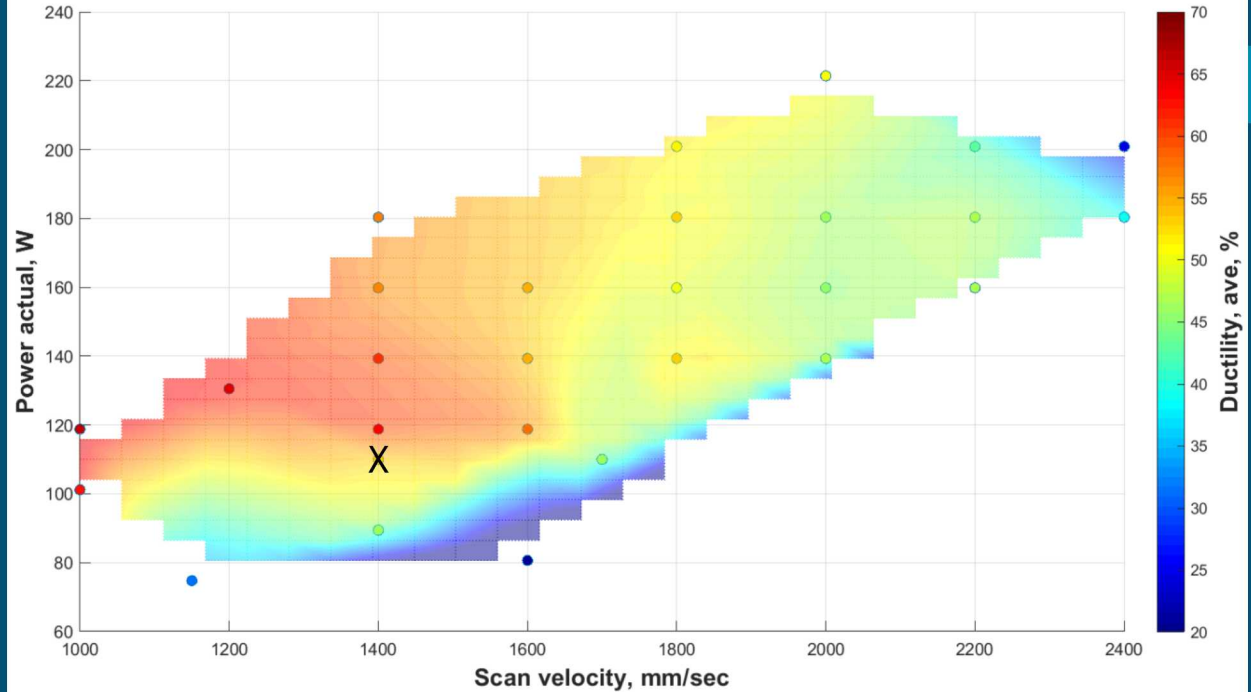
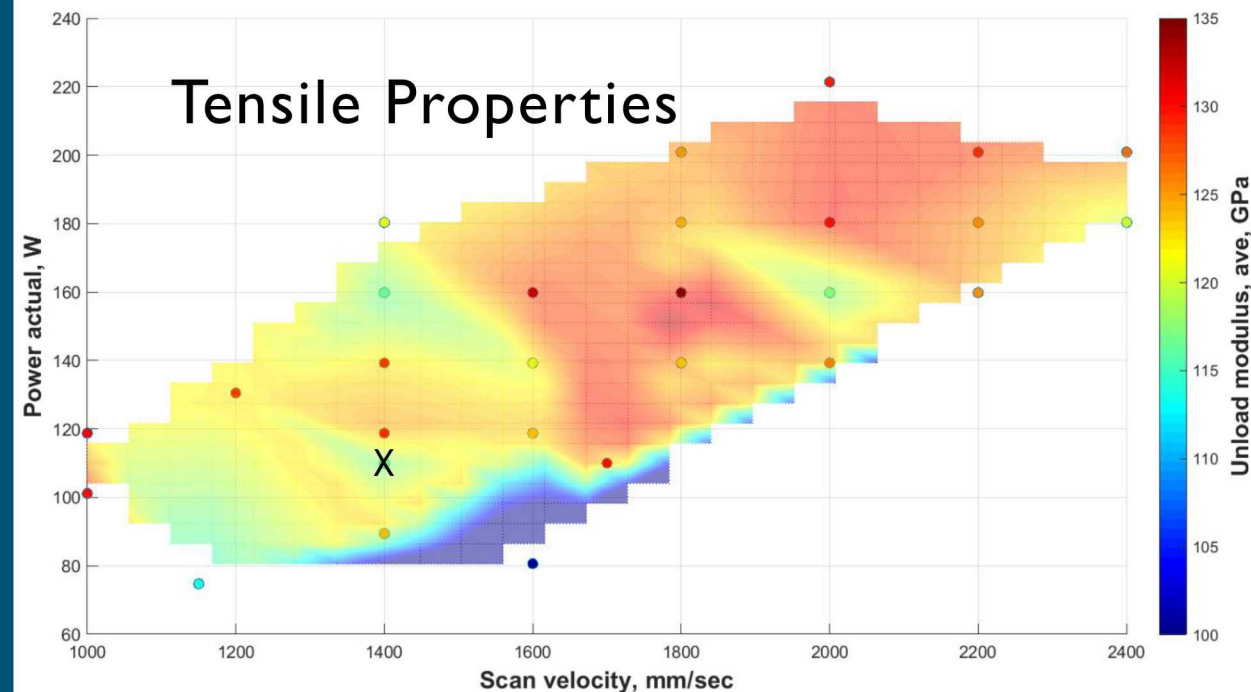


*build plate w/20  
arrays, 10  
dogbones/array*





# Tensile Properties



# Lattice Process-Structure-Properties

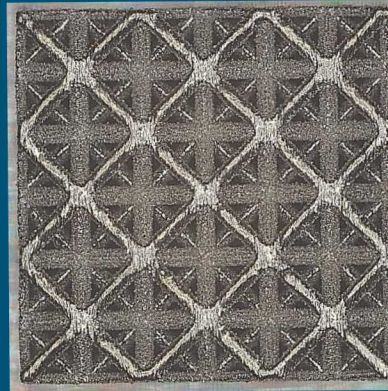
## Motivation

- qualification
- process / performance tolerances

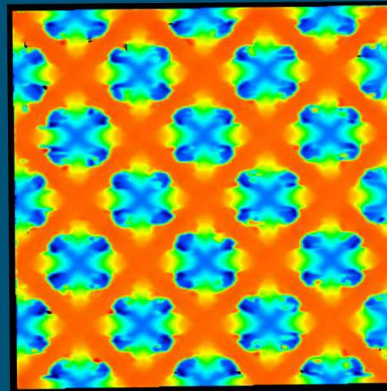
## Developing process maps

- represents additional design freedom

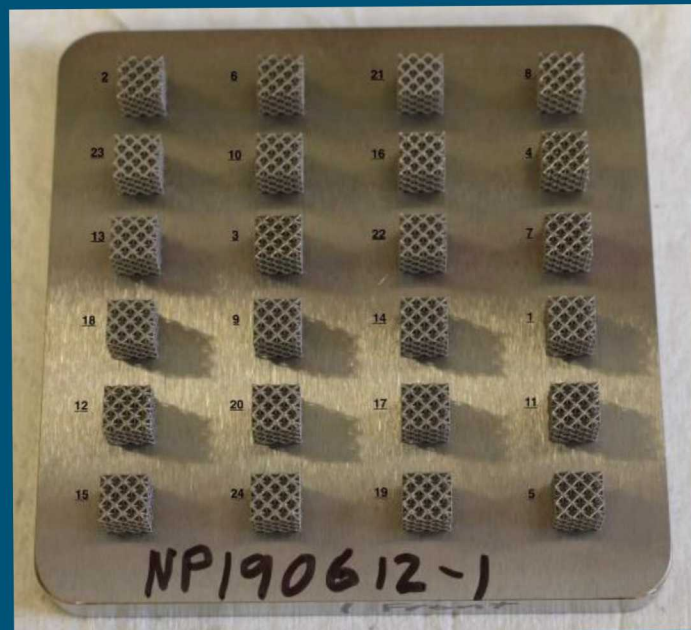
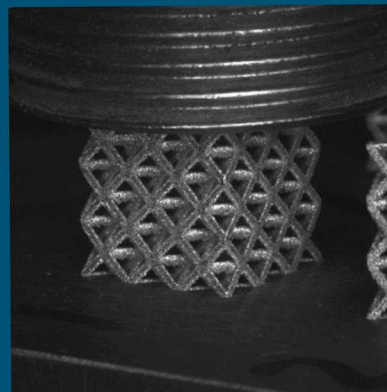
top surface  
image



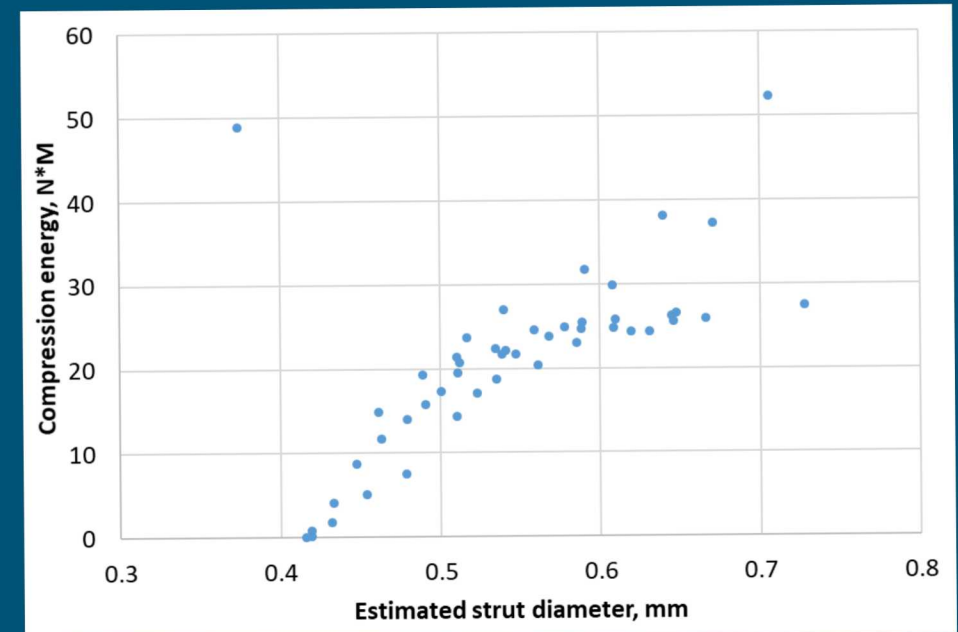
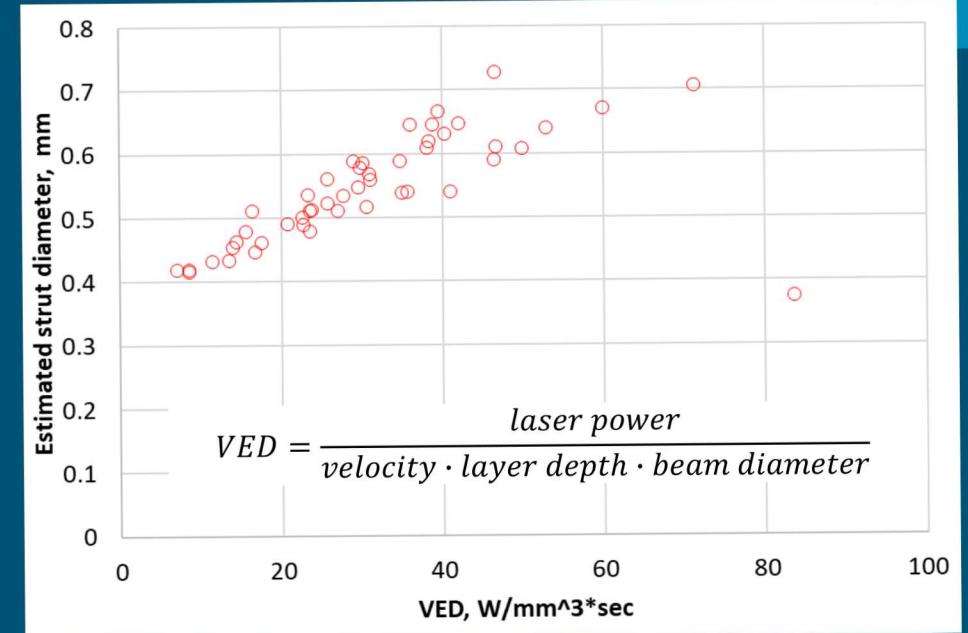
topography



compression  
testing



1cm<sup>3</sup> octet cubes build plate





# Lattice Process-Structure-Properties

## Motivation

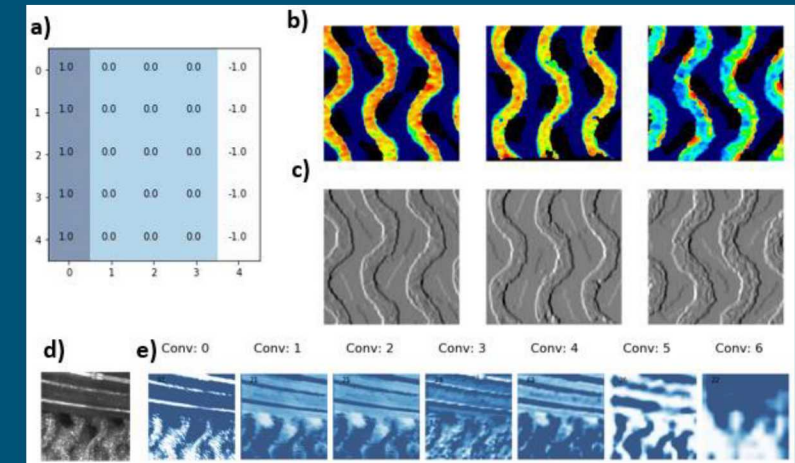
- qualification
- process / performance tolerances

## Developing process maps

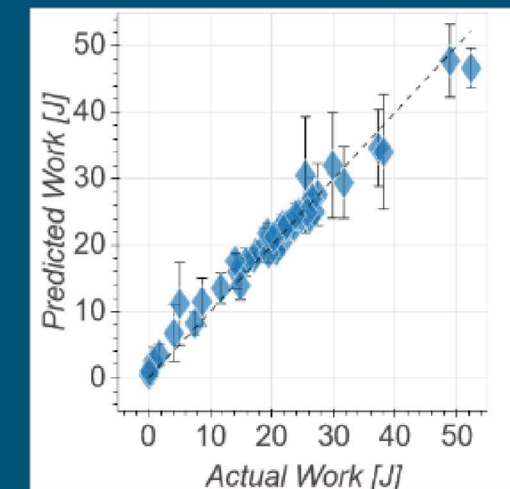
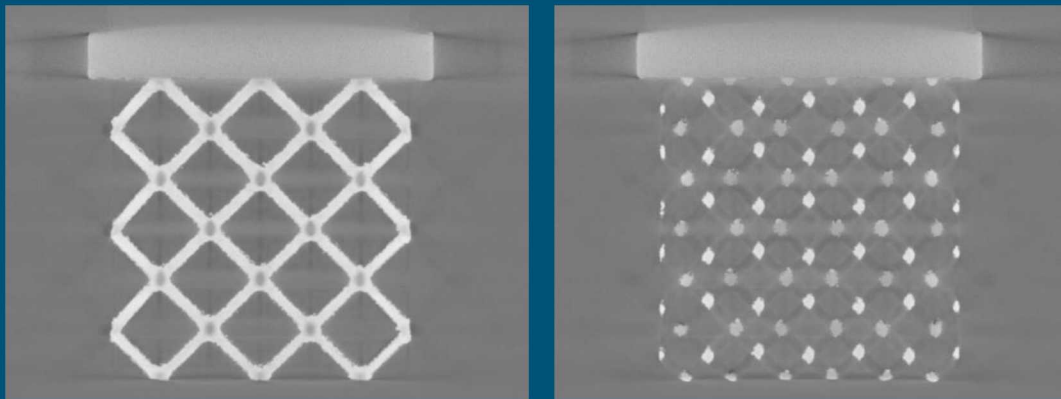
- represents additional design freedom

## Inspection

- machine learning
  - addresses “feature engineering” challenges
- high throughput computed tomography



CT images from 10min scan of octet with struts (left) & nodes (right), 12 $\mu$ m voxels



# Laser Powder Bed Fusion

## 3D Systems ProX 200

- CW ytterbium, single mode fiber laser
  - 1070nm, 300W max power, 100 $\mu$ m dia.
  - scan speed = 1.4m/sec
- rotating powder roller
- Ar backfill & cover gas flow

## Part capabilities

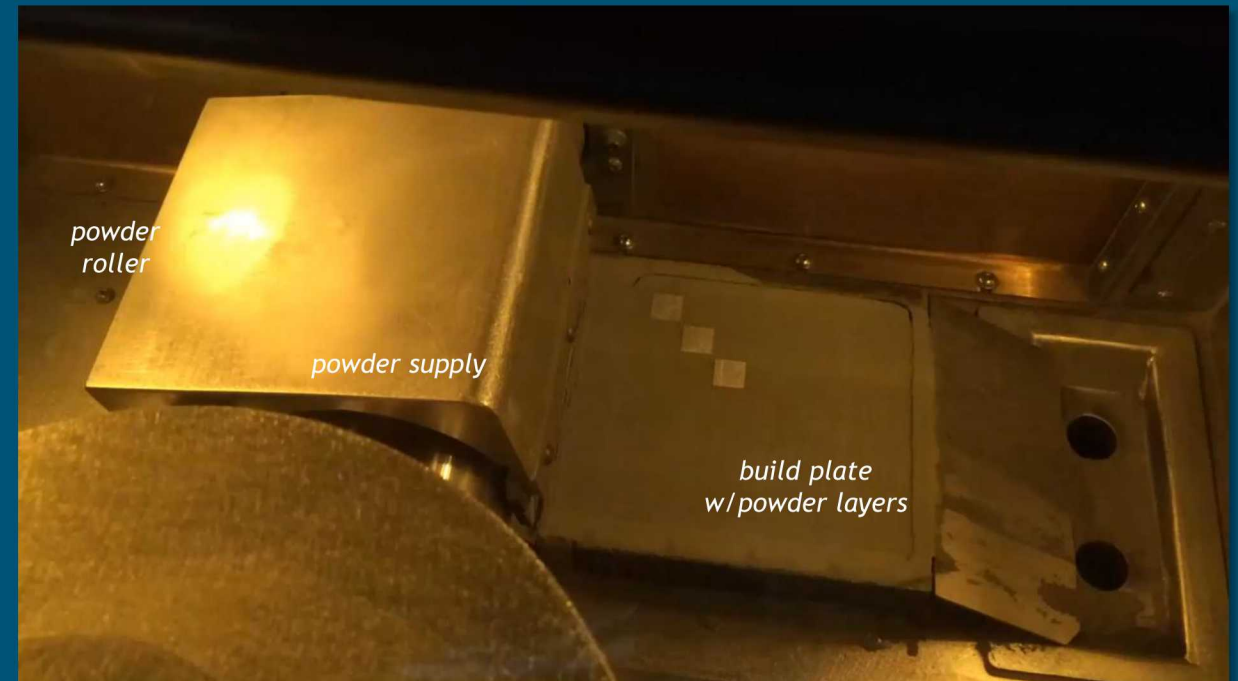
- 140x140x100mm build volume
- 316L stainless steel
  - 20-30 $\mu$ m diameter powder
- deposition rate  $\sim 100\text{mm}^3/\text{min}$
- $10^5\text{-}10^6\text{C/s}$  heating & cooling rates



high throughput  
dogbone sample



Sandia T-  
bird lattice





# Motion Stage Errors

## 100mm ball screw drives

- supply piston – controls powder dosage
- build piston – controls layer thickness

## Measurements

- LJ-V7200 laser profiler
  - range:  $\pm 48\text{mm}$
  - Z repeatability:  $1\mu\text{m}$

## Replicate build process

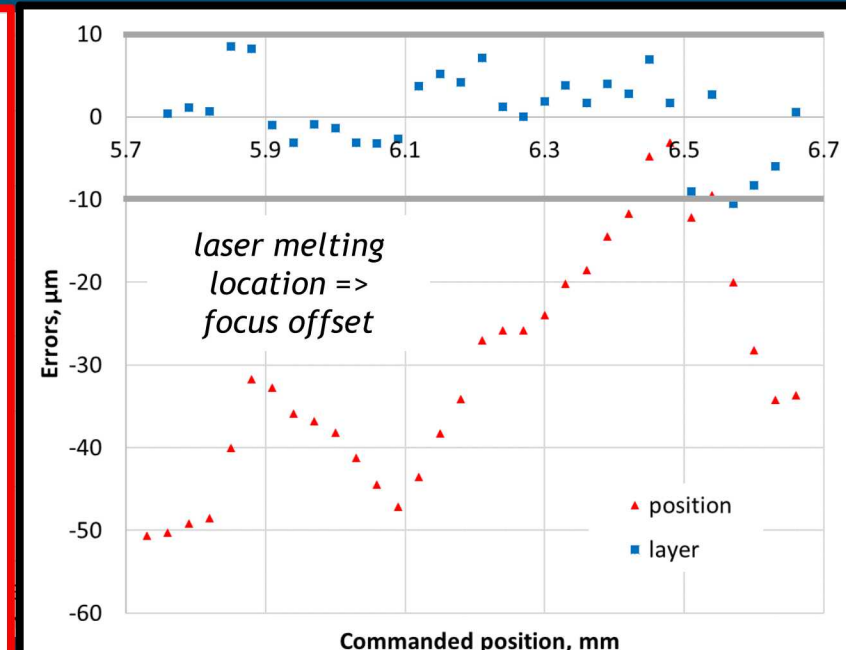
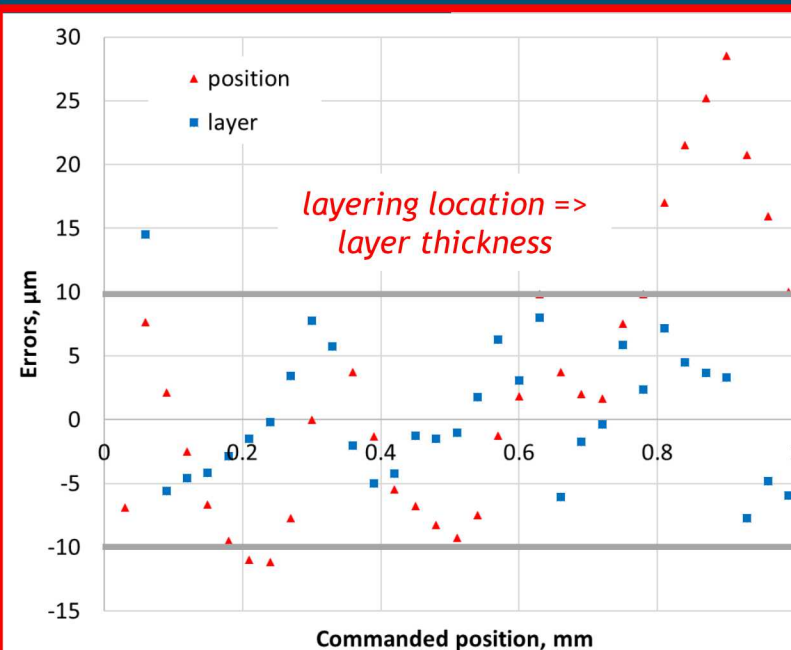
- $30\mu\text{m}$  layers
  - layer then laser melting locations
- step errors  $\sim$  powder size
  - problem?
  - process exhibits some “self-healing”



supply piston drive (left) & build piston drive (right)



Keyence LJ-V7200 focused onto build plate

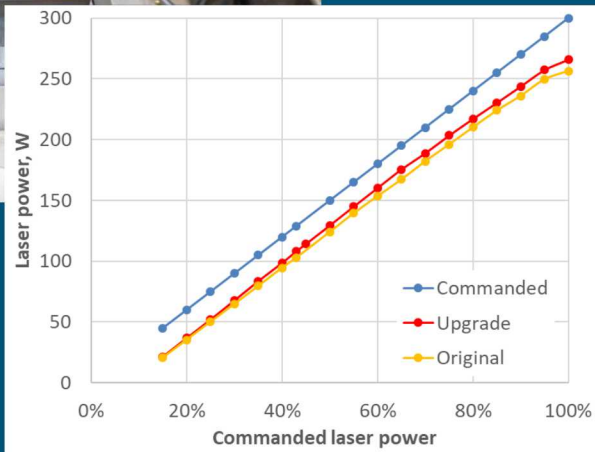
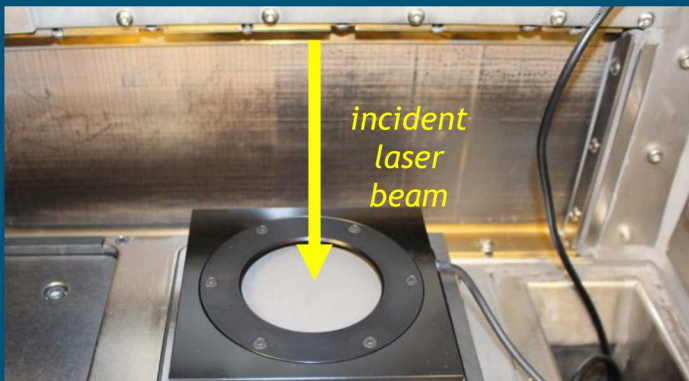


# Laser Characterization

Power @ powder layer, not machine setting

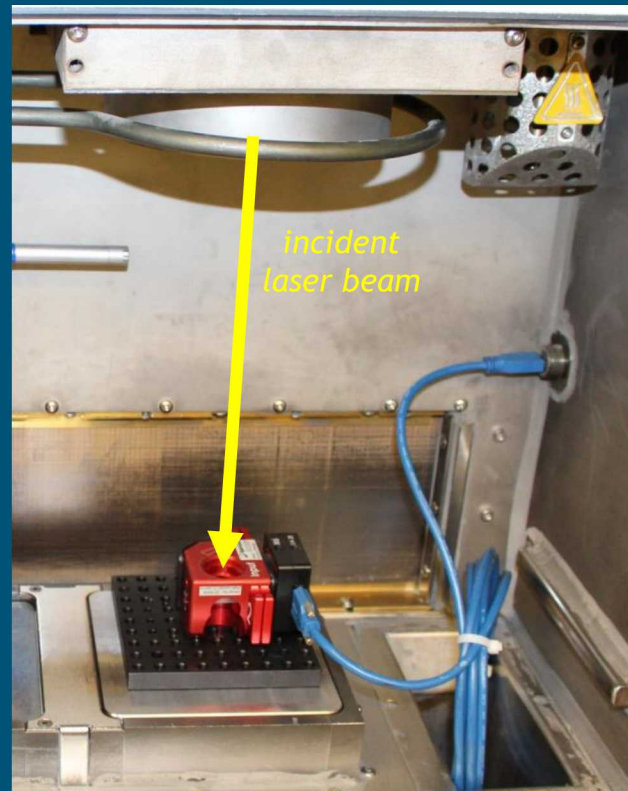
- required for accurate & repeatable process optimization

*Ophir L50(300)A-LP1 power meter on the build plate*

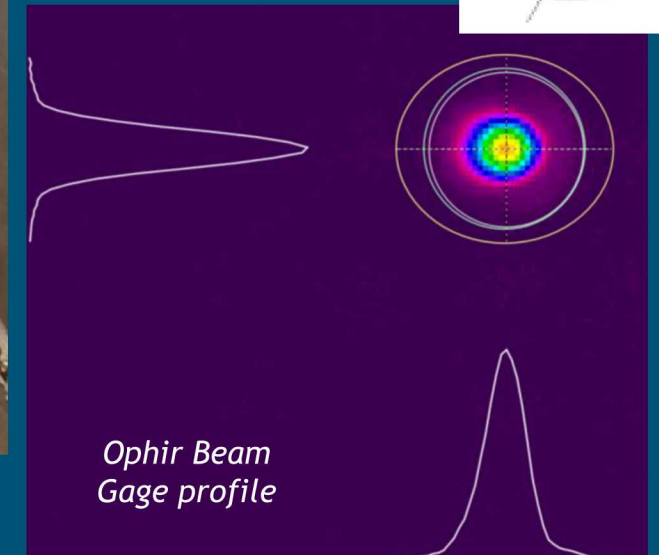
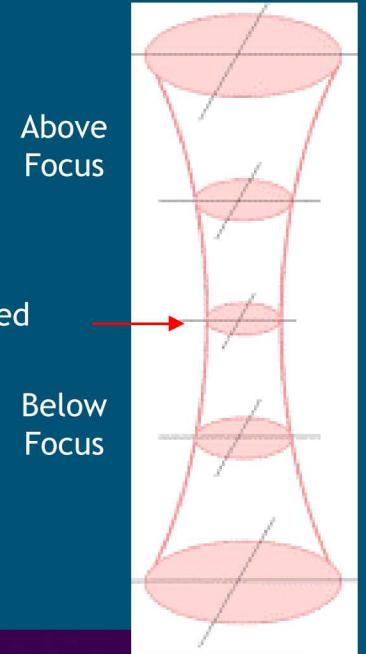


## Beam profiling

- diameter along optical axis
- variation w/laser power
- temporal response
- heating & cooling
- original & upgraded f-theta optics



*Ophir SP 928 profiler on the build plate*

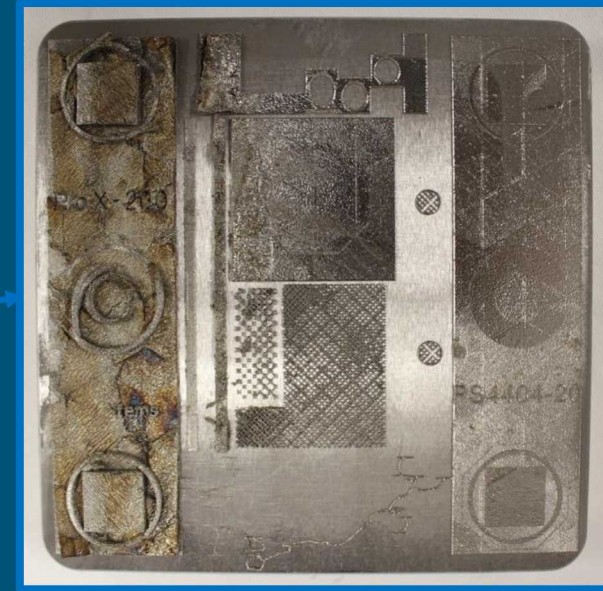
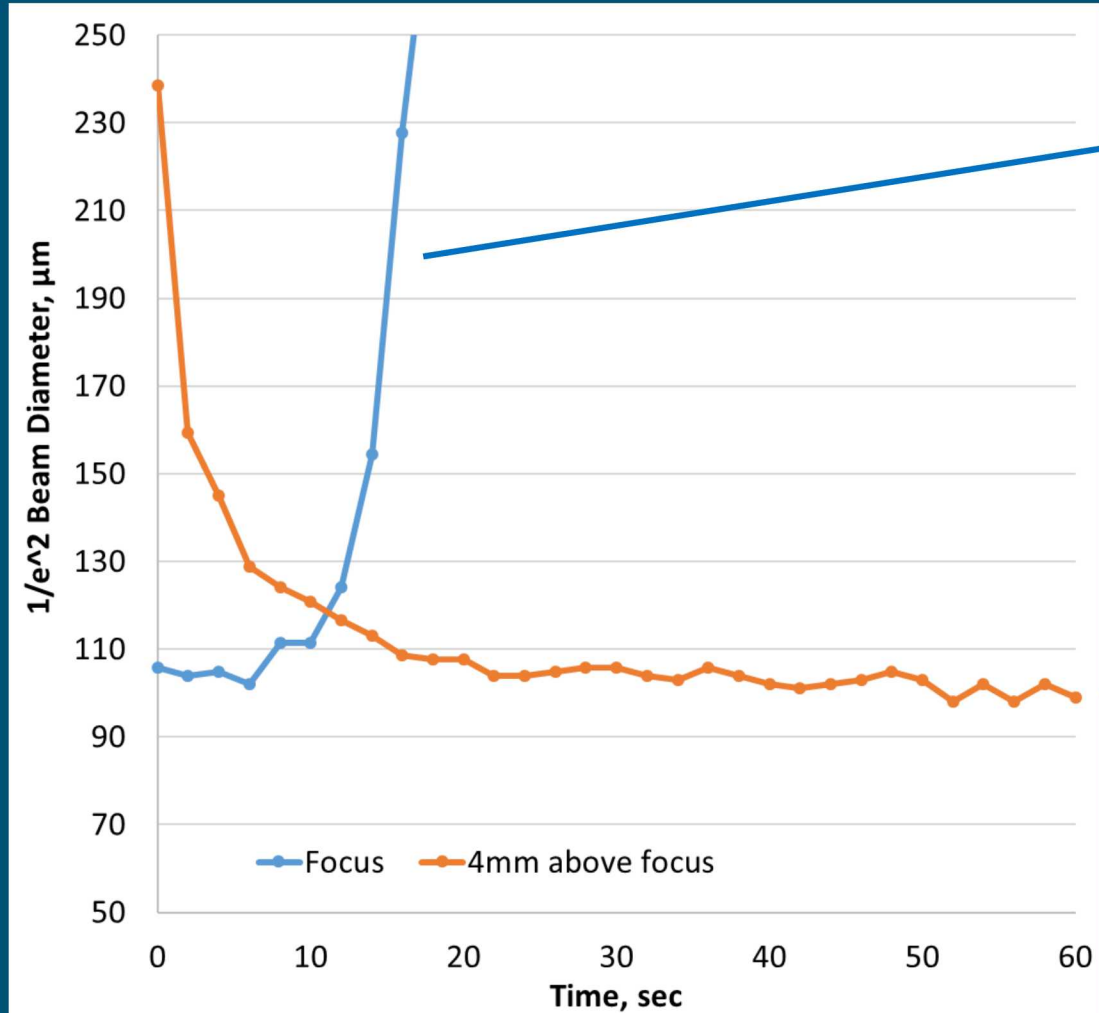




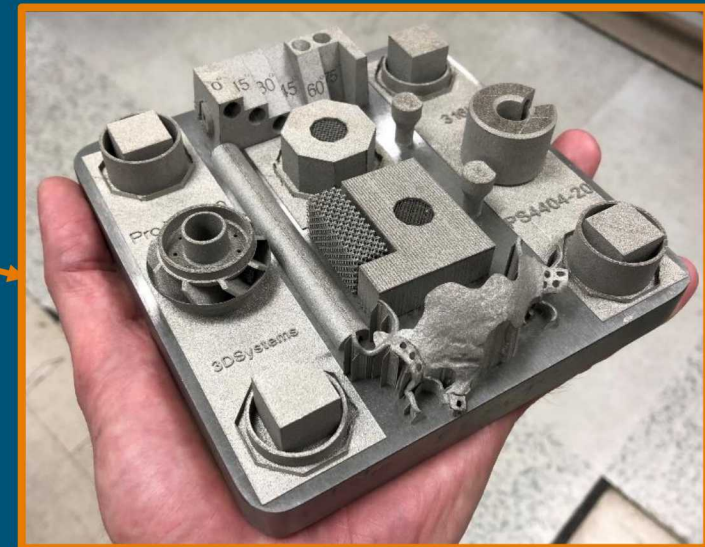
# Original F-Theta Diameter Variation

Thermal lensing obscures optimal focus for a realistic build cycle

- introduces challenges & uncertainties to part & material generation



*nominal focus offset*



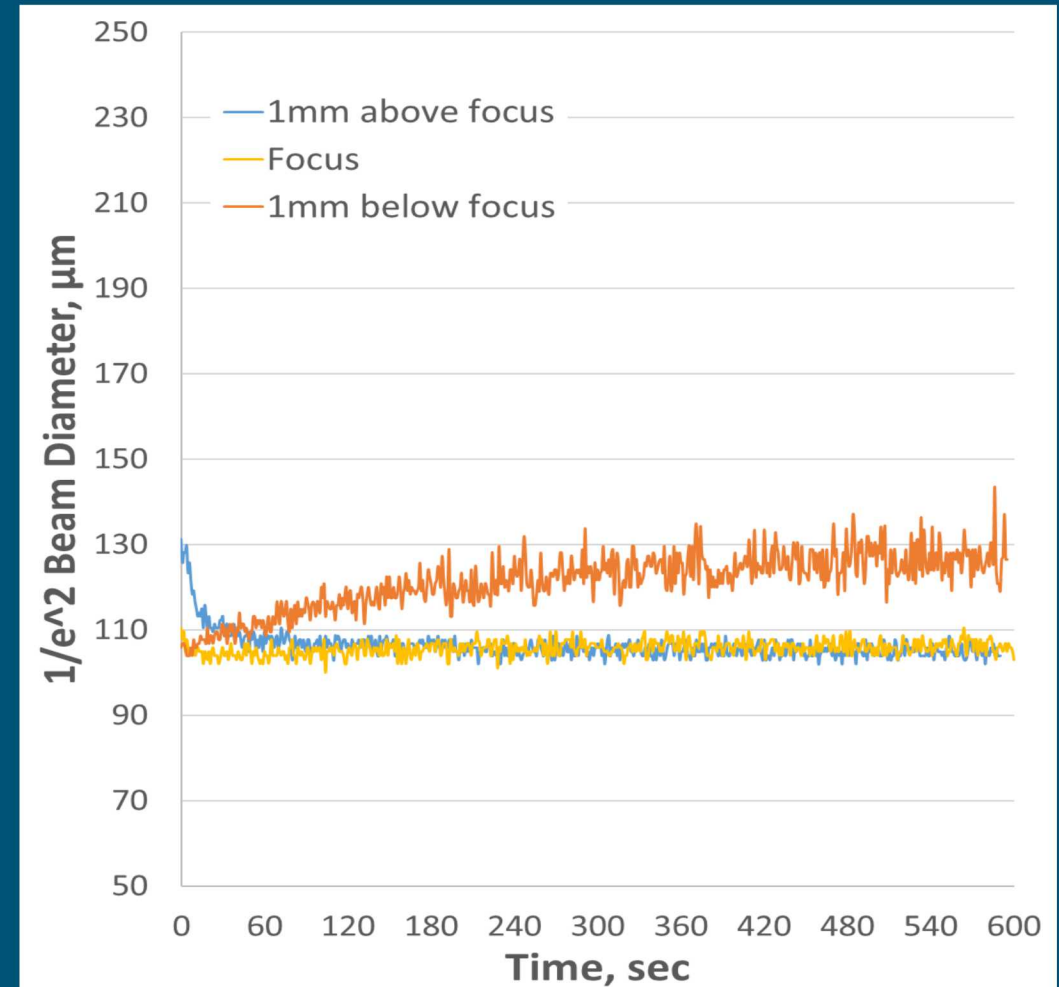
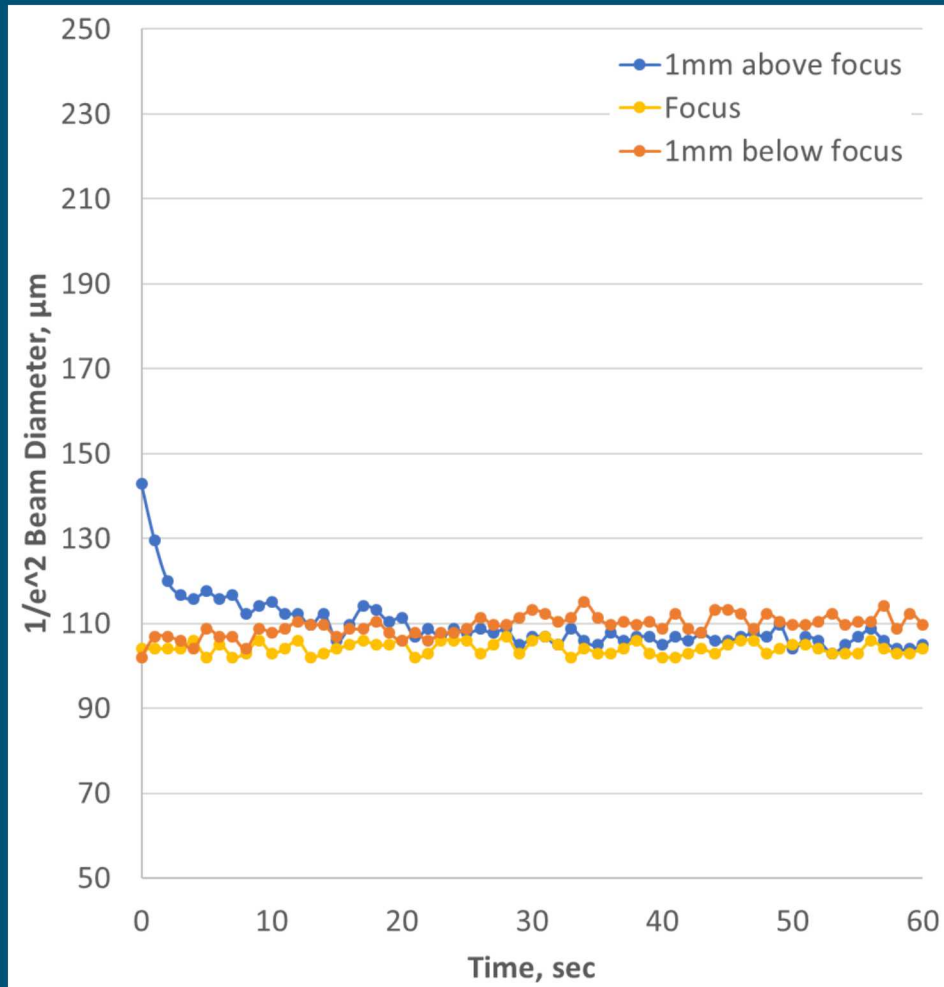
*4mm focus offset*

## Upgraded F-Theta Diameter Variation



Dramatic reduction in thermal lensing & improved repeatability / consistency

Critical to process optimization efforts

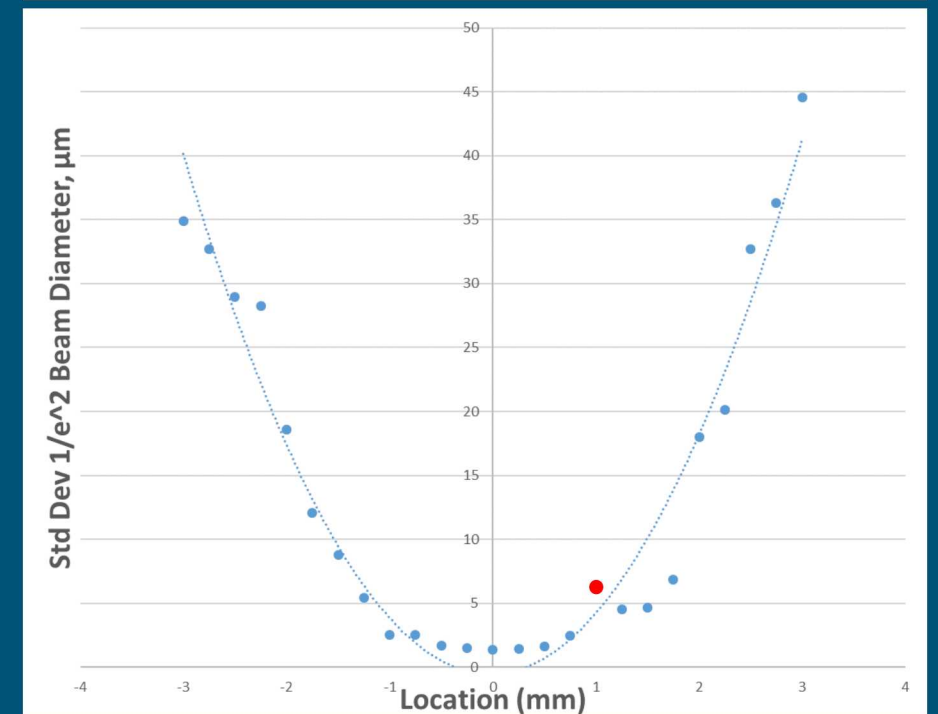
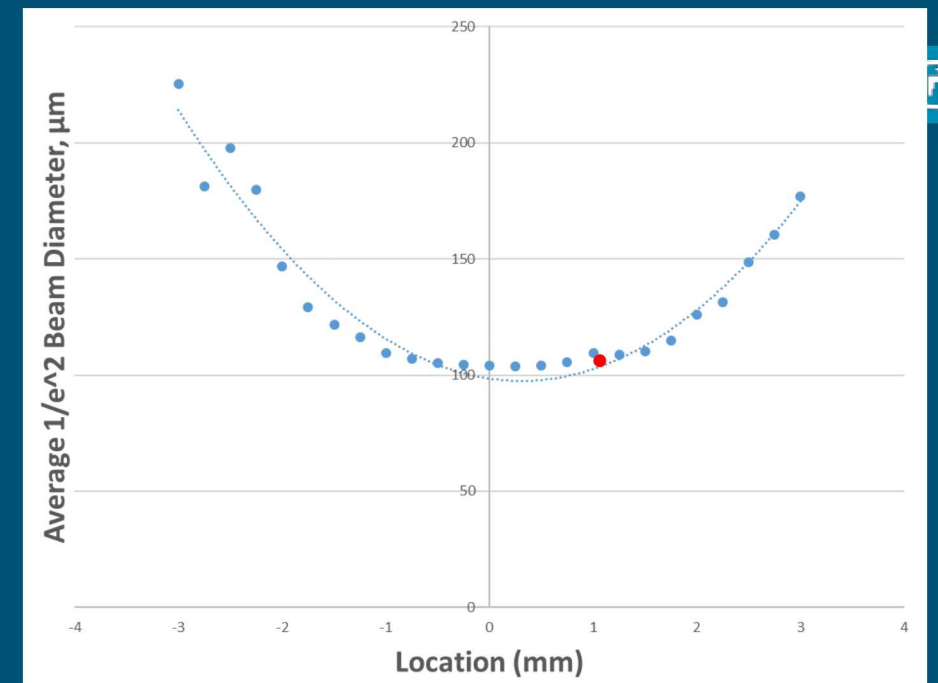
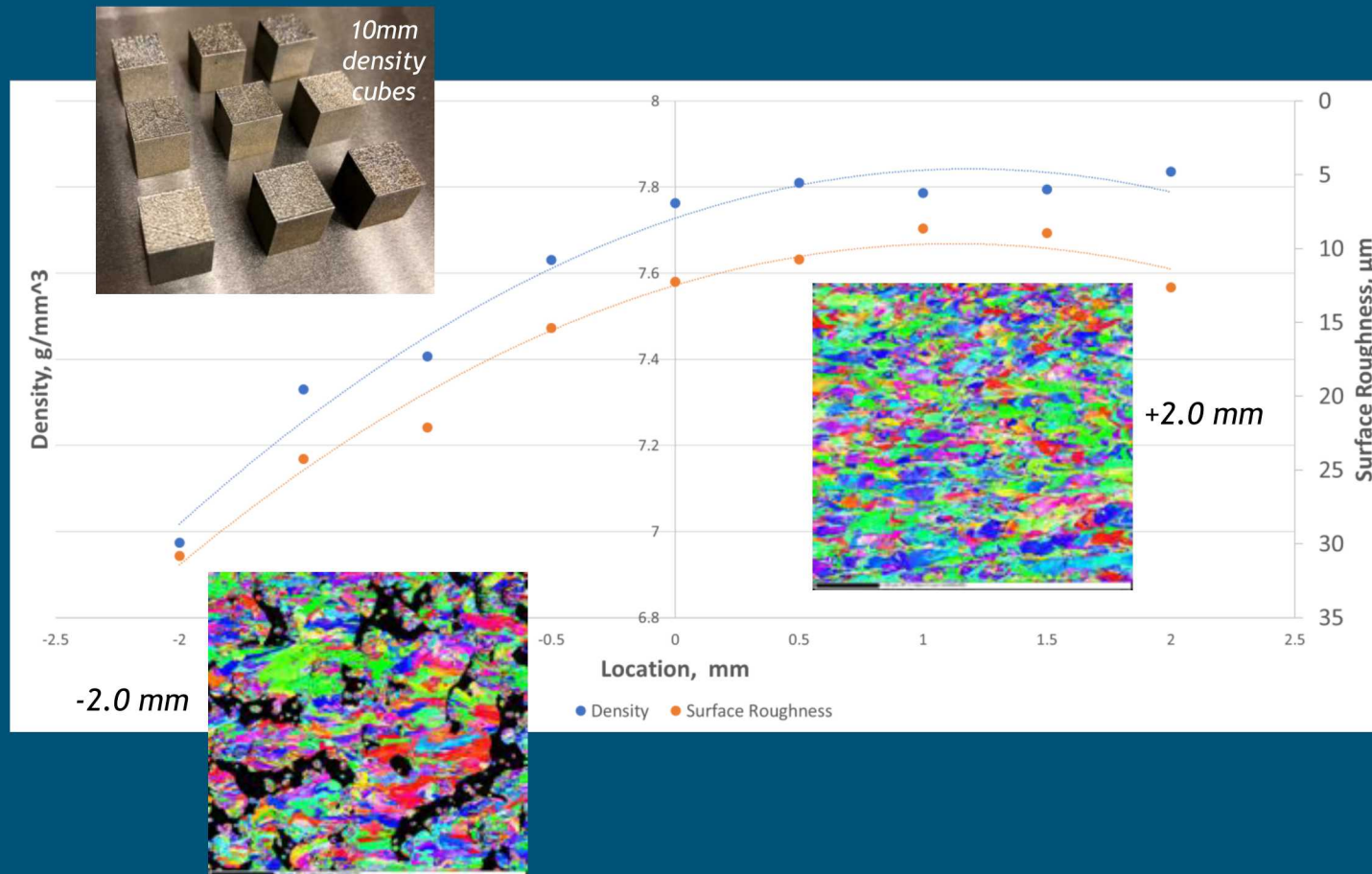


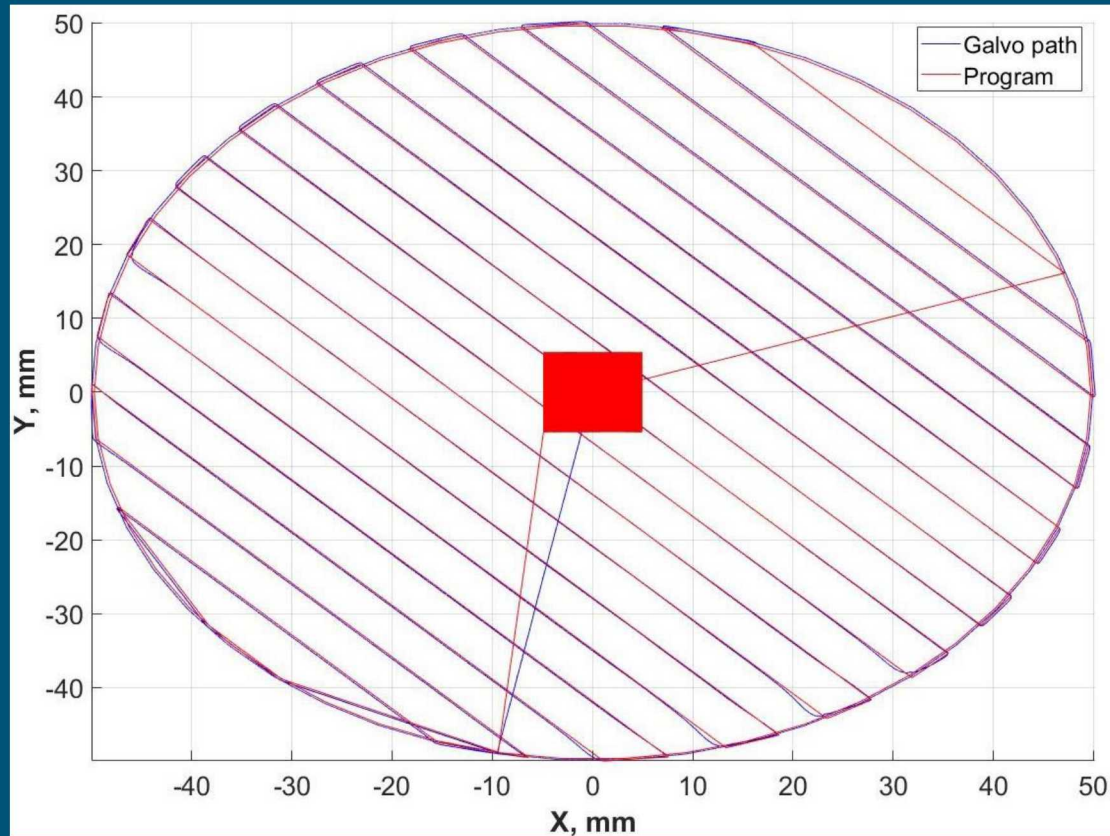


# Upgraded F-Theta Focal Distance

How critical is build plate location relative to laser focus?

- “optimal” process appears to be +1mm from “focus”
- presumably plasma plume impacts optimal offset



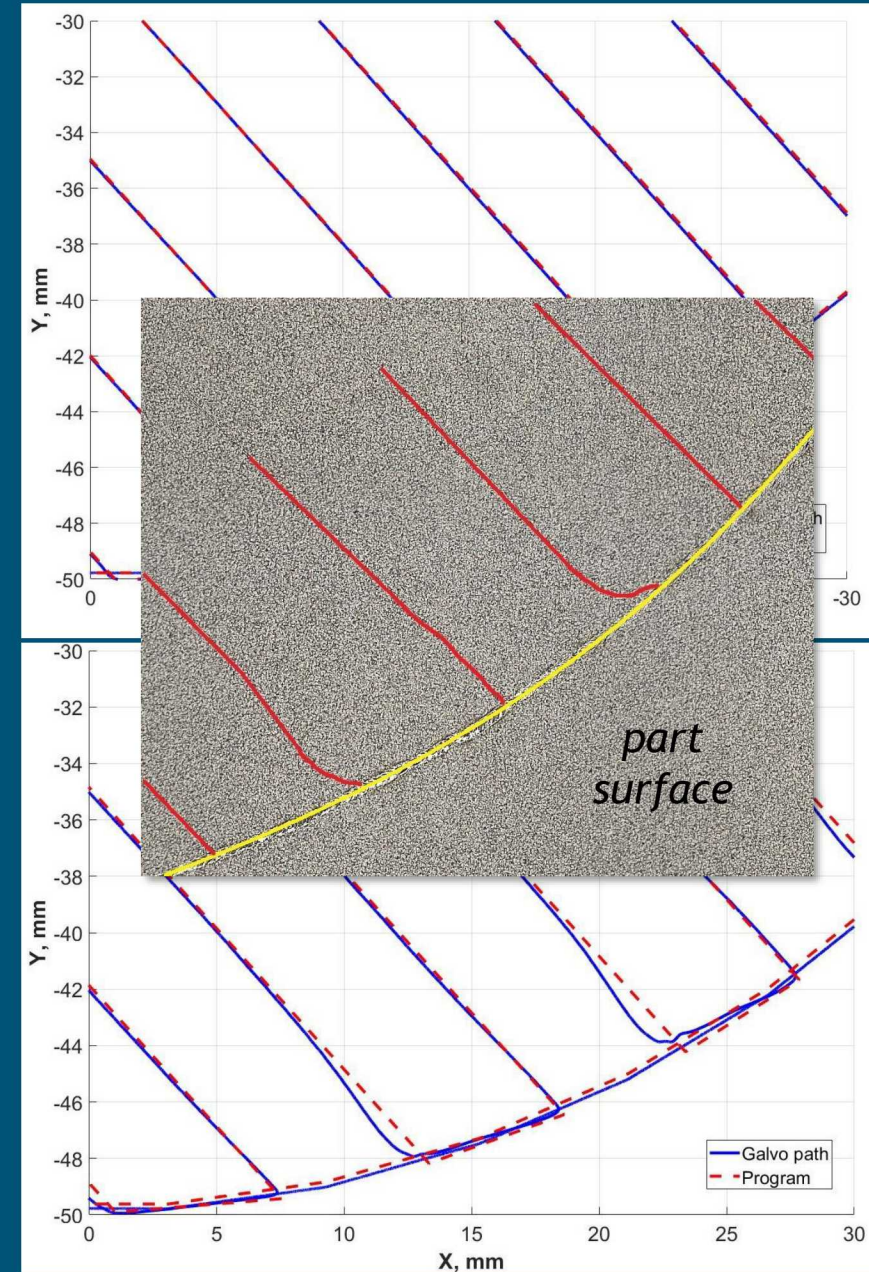


ARCS output for a simple test pattern

- Archive, Research, Control, Synchronization (ARCS)
  - Penn State, 3D Systems collaboration
  - motion @ 100kHz

*desired  
motion*

*galvo  
errors*





Material formation concurrent  
w/ geometry

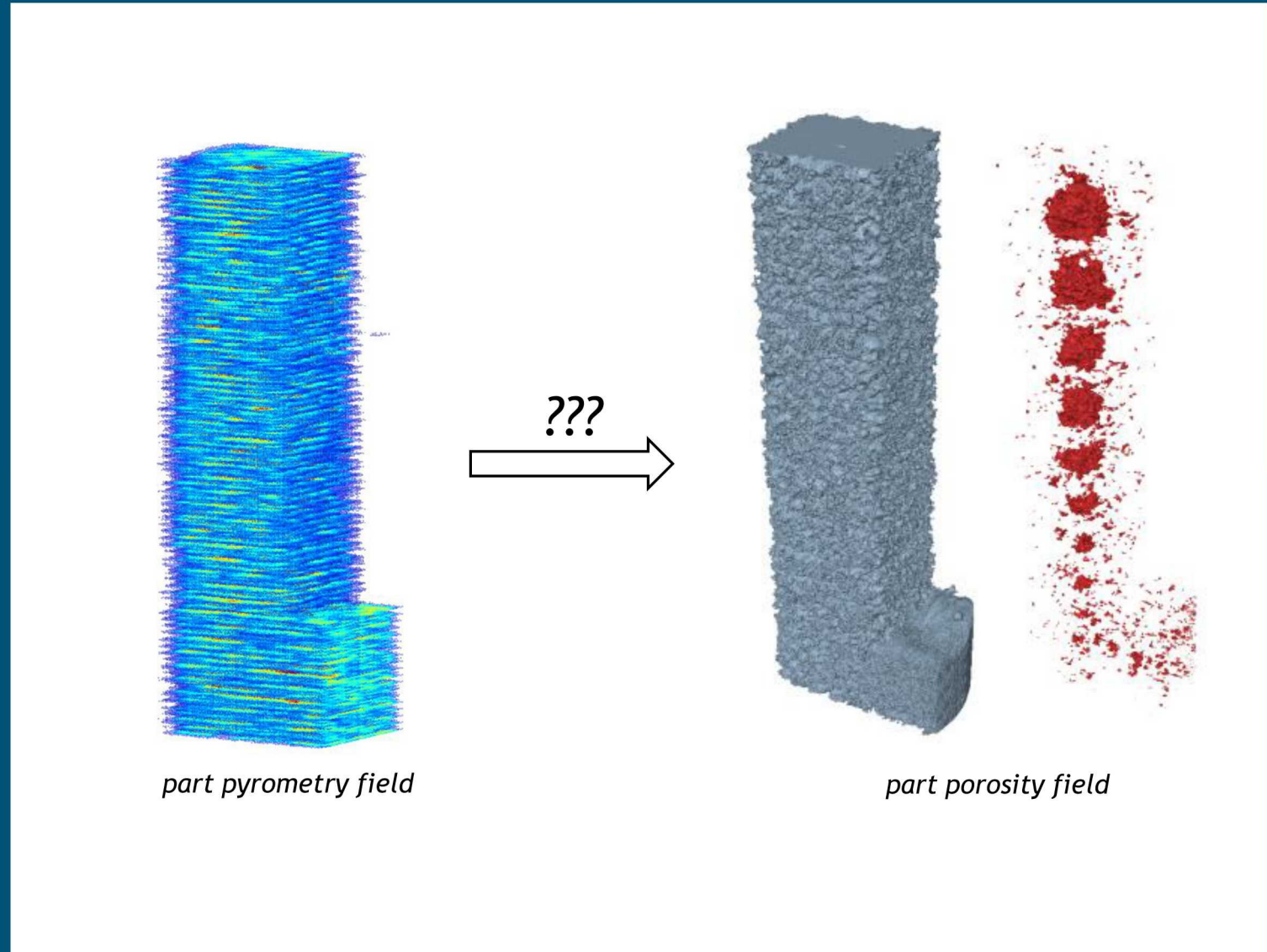
- want to predict part/material performance
- qualification / acceptance requires
  - significant design margins
  - understanding defect impacts on performance

How to ID a bad part?

- proof testing
- non-destructive inspection
- *process monitoring*

Objective

- correlate sensor data (X,Y,Z,time) to spatial material porosity (X,Y,Z)



# There are Multiple Potential Melt Pool Signatures for Interrogation



## Thermal

- Stratronics ThermoViz two-color pyrometer
- IR cameras: FLIR C2, A310 & SC6811

## Optical

- Photron PhotoCam Speeder V2 high speed cameras
- blue light illumination
- Ocean Optics LIBS2500plus spectrometer
- Keyence LJ-V7020 & LJ-V7200 laser line scanners

## Acoustic

- audio microphone, acoustic emission

## 3D Systems Open Protocol platform

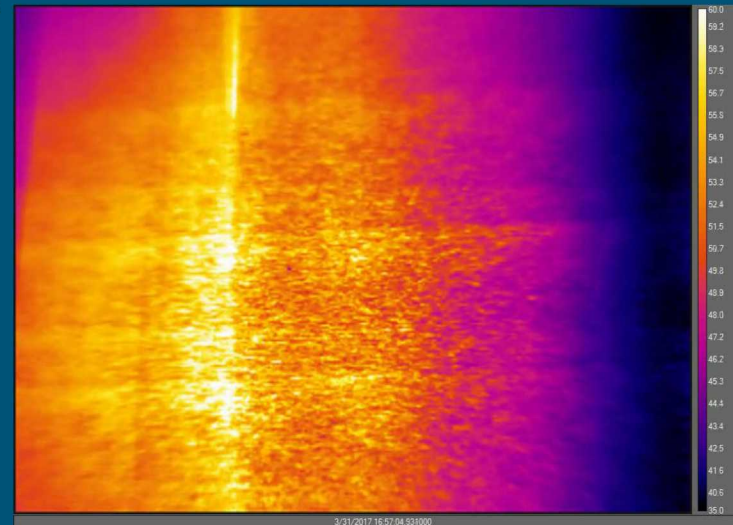
- PSU multi-spectral sensor

## Managing & analyzing data streams is crucial

- large
- non-linear correlations



*ThermaViz installed in the ProX 200*



*FLIR A310, laser on plate, ~100W, 1.4m/sec, 125μm hatch, 100μm beam dia.*



*Photron high speed optical melt pool video*

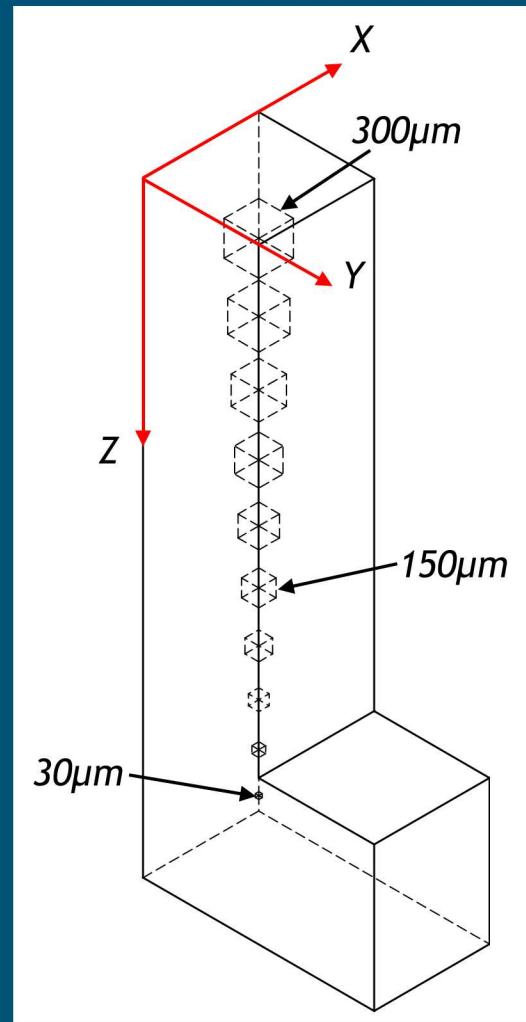


### Design

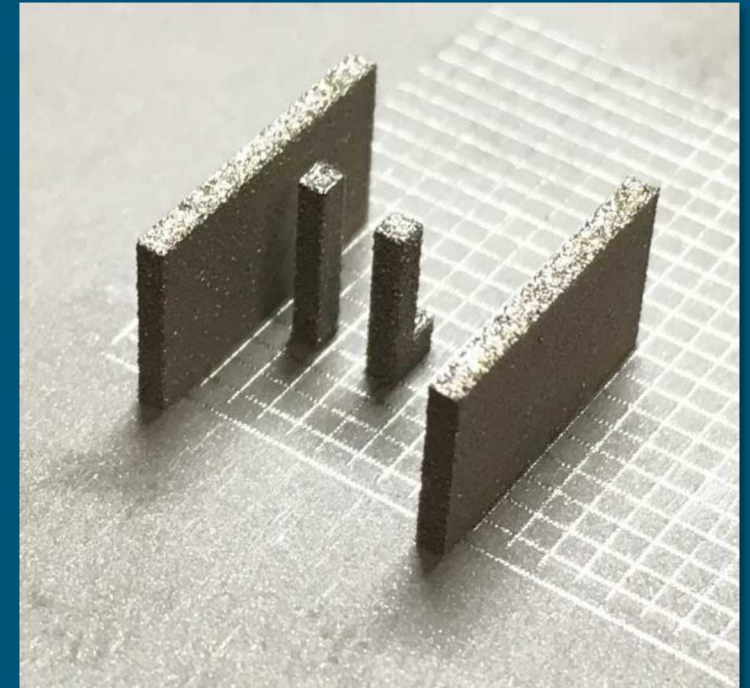
- 1x1x5.25mm column, 175 layers
- holes
  - 1-10 layer thickness (30-300 $\mu$ m cubes)
- 316L stainless steel

### Material characterization

- Zeiss Xradia 520 Versa  $\mu$ CT
- voxel resolution: **1.98 $\mu$ m**



*captured holes design*



*captured holes part printed in 316LSS*



part side view



micro-computed tomography

Adobe Photoshop



batch processing  
16bit > 8bit conversion  
lossless filetype conversion



alignment & registration  
cropping  
grayscale matching  
autoleveling  
image filtering  
thresholding



Interactive Data Language

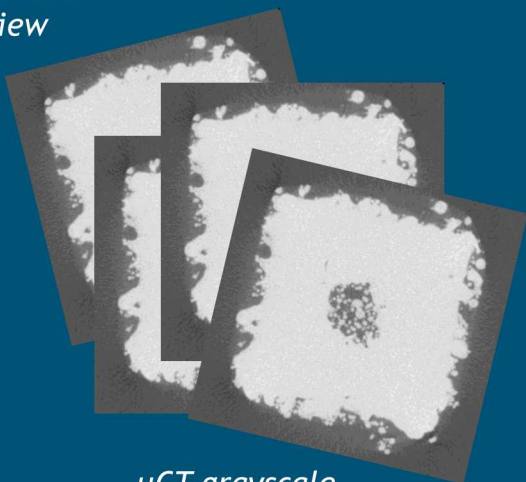


reconstruction  
quantification



Parallel Visualization Application

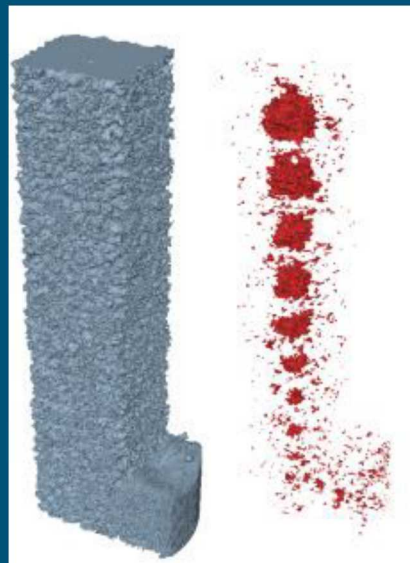
image processing  
visualization



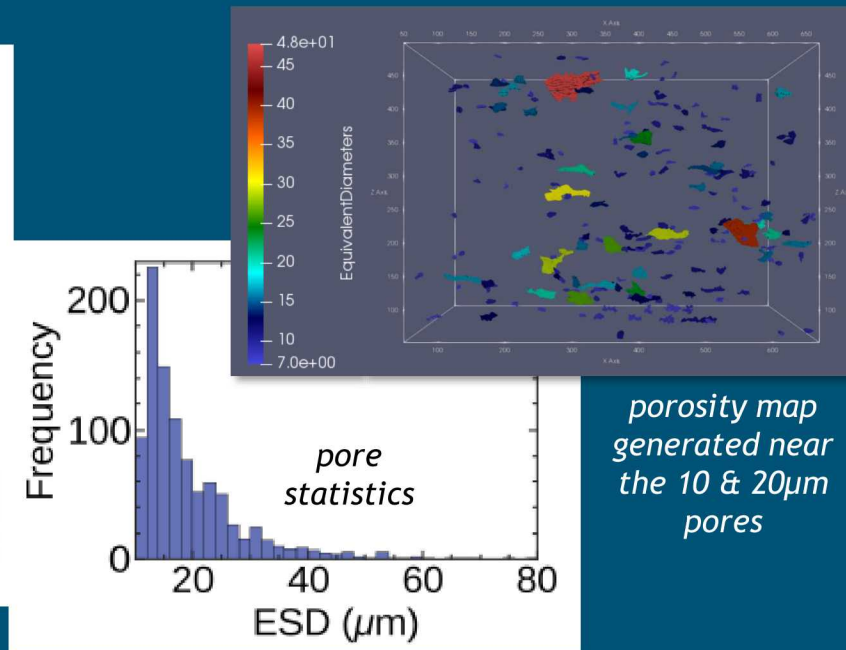
μCT grayscale image stack



aligned & segmented bitmap stack  
white = material, black = air/void



part reconstruction



porosity map generated near the 10 & 20μm pores



# Melt Pool Monitoring

## Stratronics Therma-Viz 2-color pyrometer

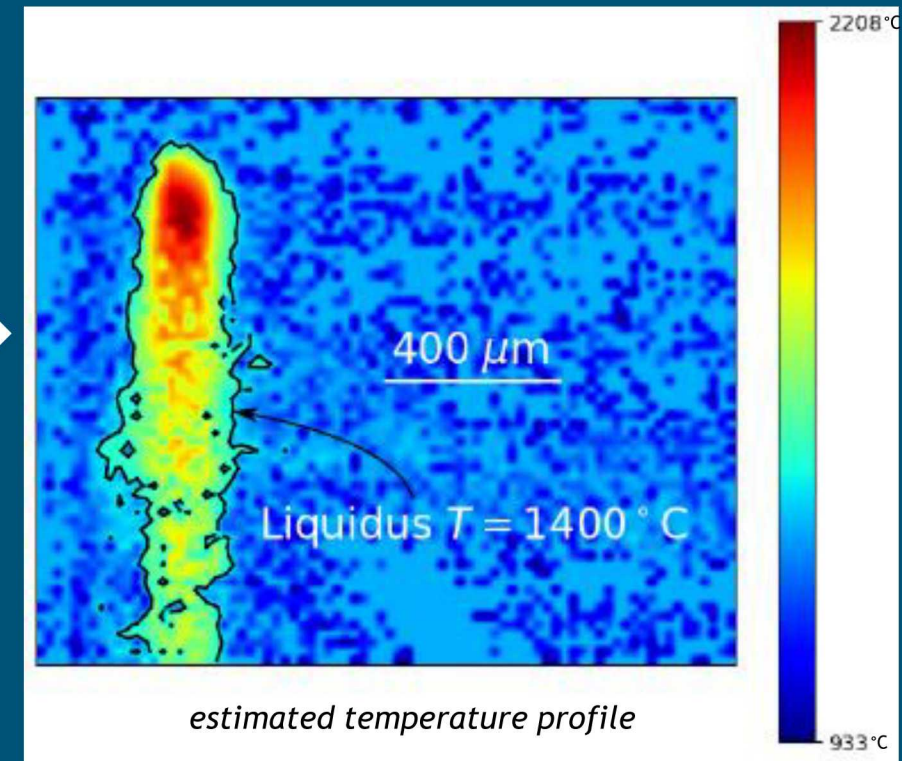
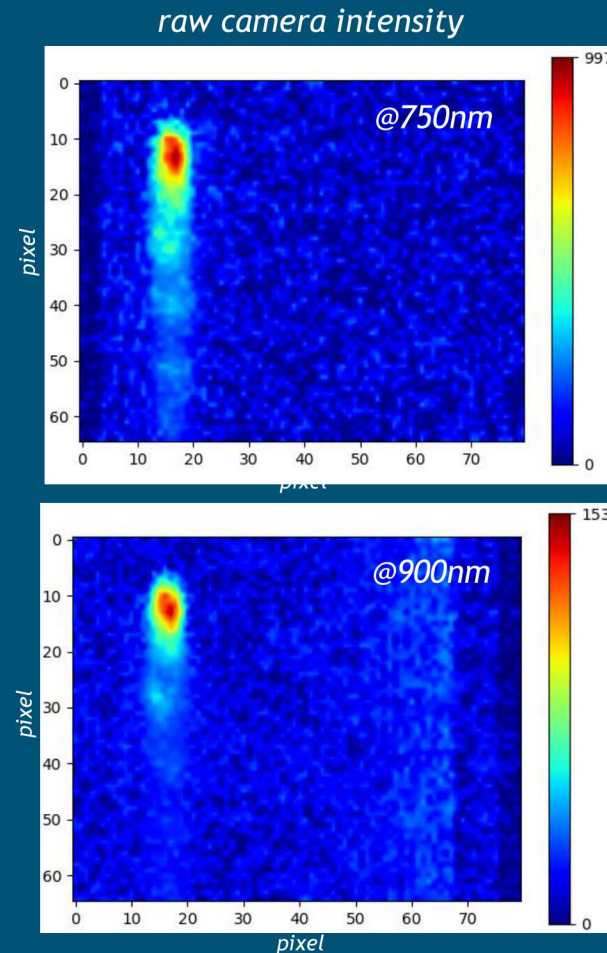
- CMOS imagers:  $\sim 20\mu\text{m}/\text{pixel}$
- 750 & 900nm filters
  - 50nm bandpass
- $\text{Temp}_{\text{pixel}} \cong \text{func}(I_{750\text{nm}}/I_{900\text{nm}})$

## Fixed field, angled viewing

- FOV: 80 x 65 pixels (1.6 x 1.3mm)
- frame rate: 6-7kHz
  - desire >10kHz based on scan velocity

## Data analysis

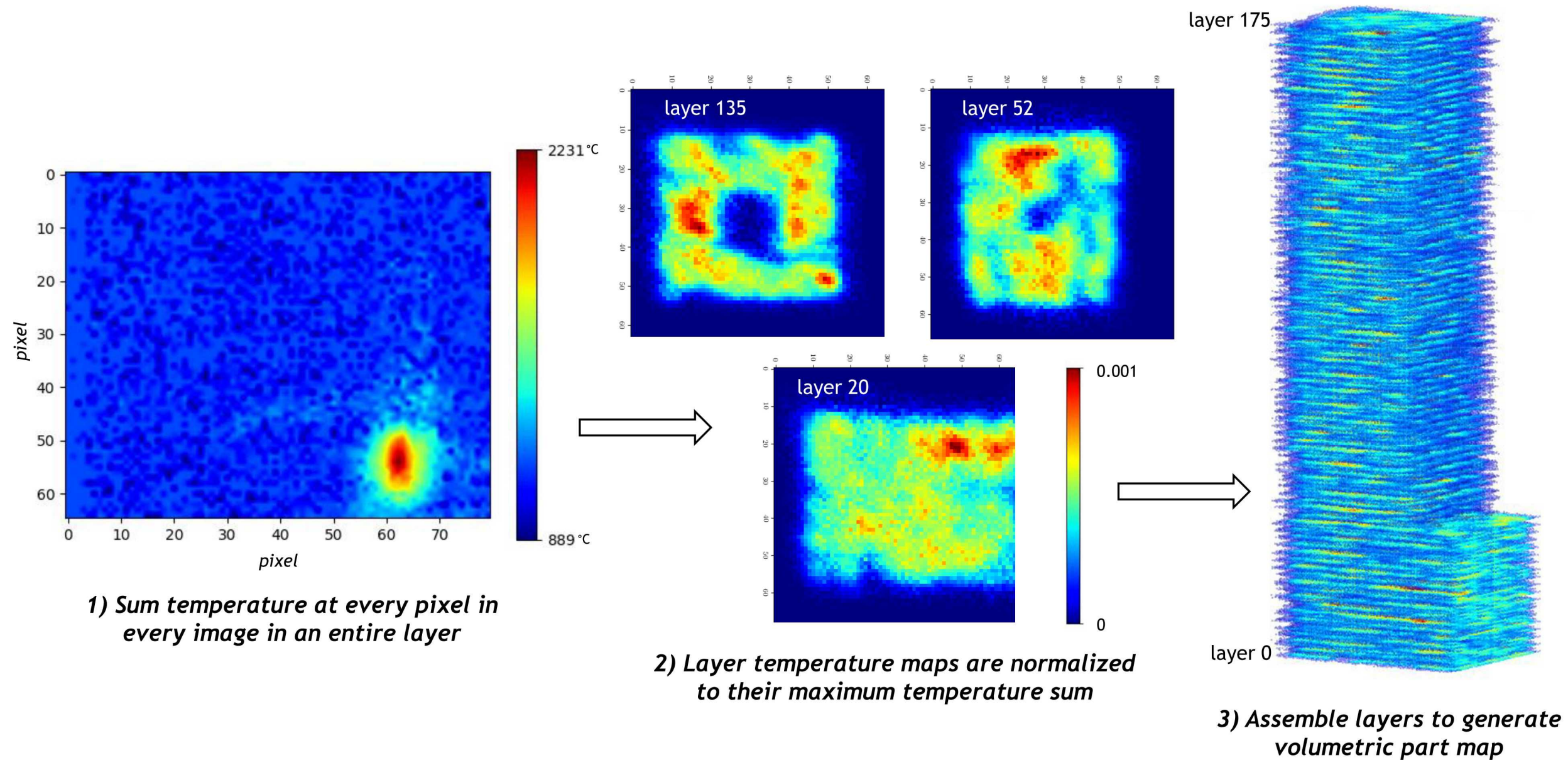
- 87,500 image frames per part
- custom scripts to estimate temperature & melt pool characteristics



*ThermaViz installed in the 3D Systems ProX 200*









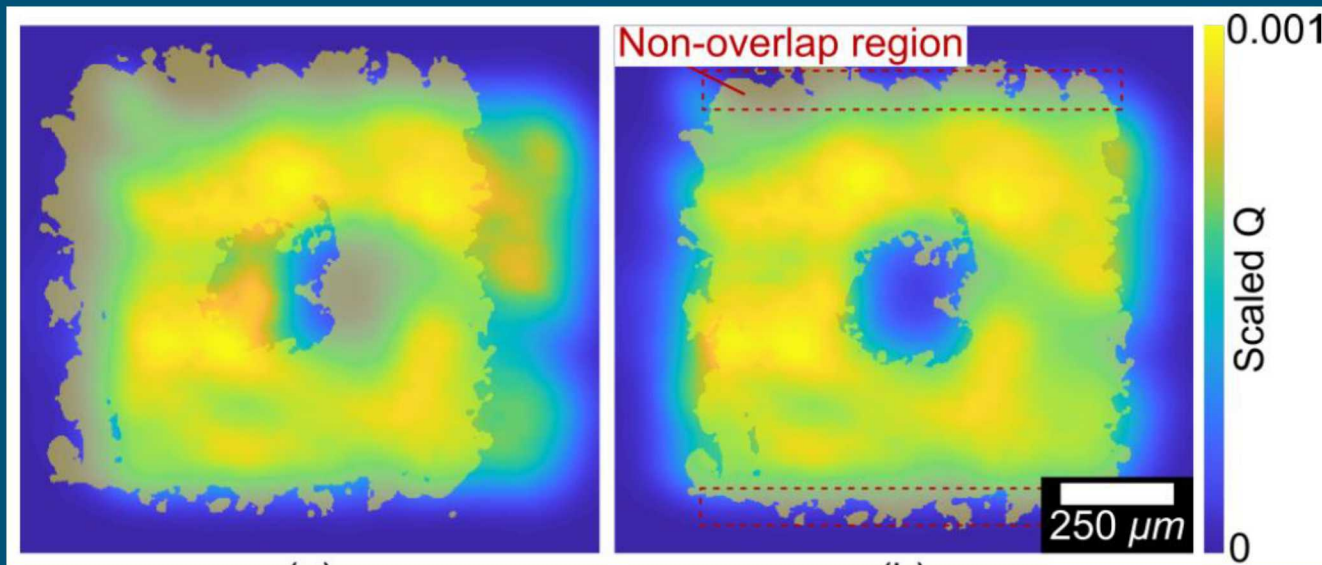
# Data Registration

Disparate spatial resolutions & aspect ratios

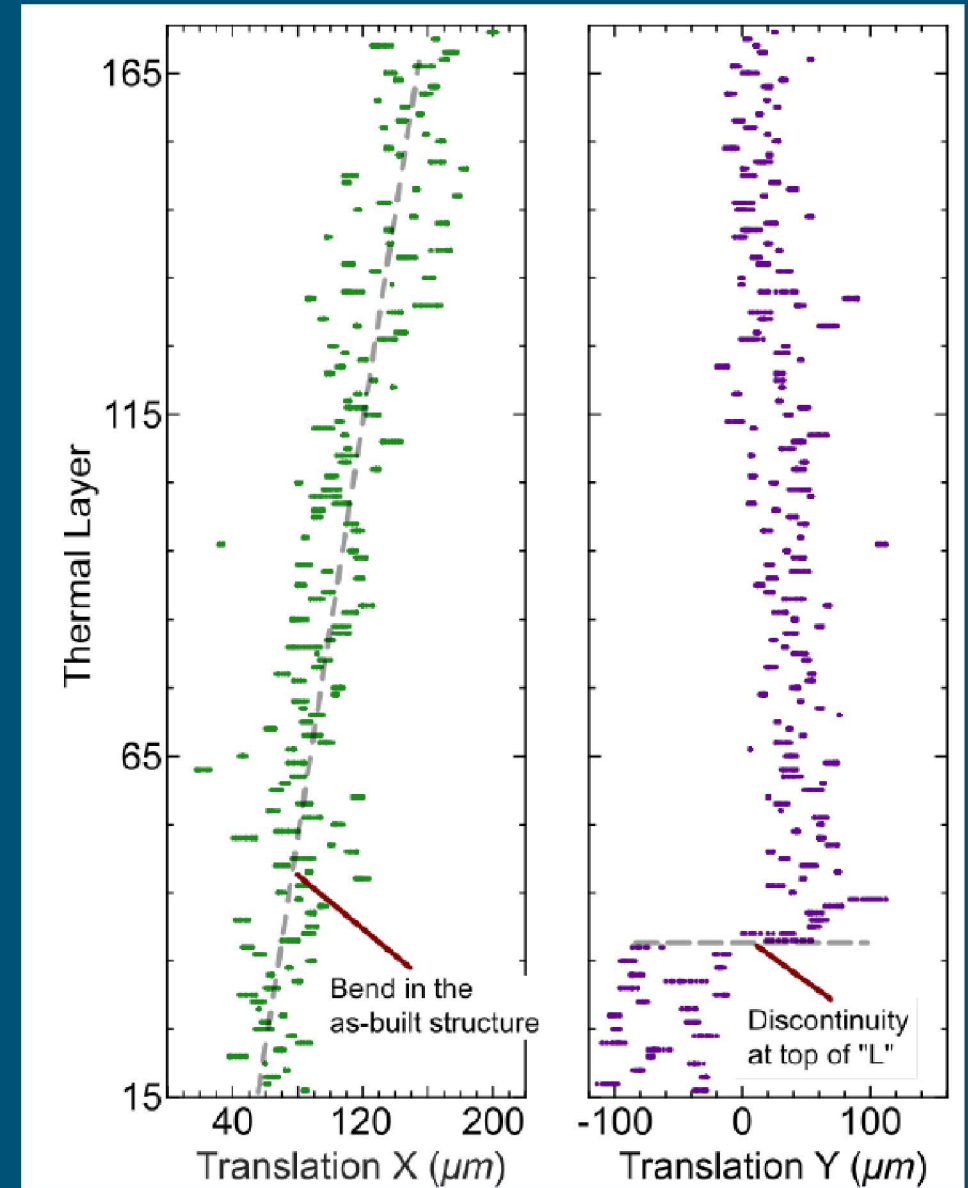
- layerwise pyrometer data scaled to  $\mu$ CT using bilinear interpolation

Part distortion

- requires linear translations in X & Y of pyrometer data
- top surface is  $Z = 0$  since EDM removes bottom geometry



overlap error near part top (left) improved after registration (right)



# Identifying Melt Pool Outliers

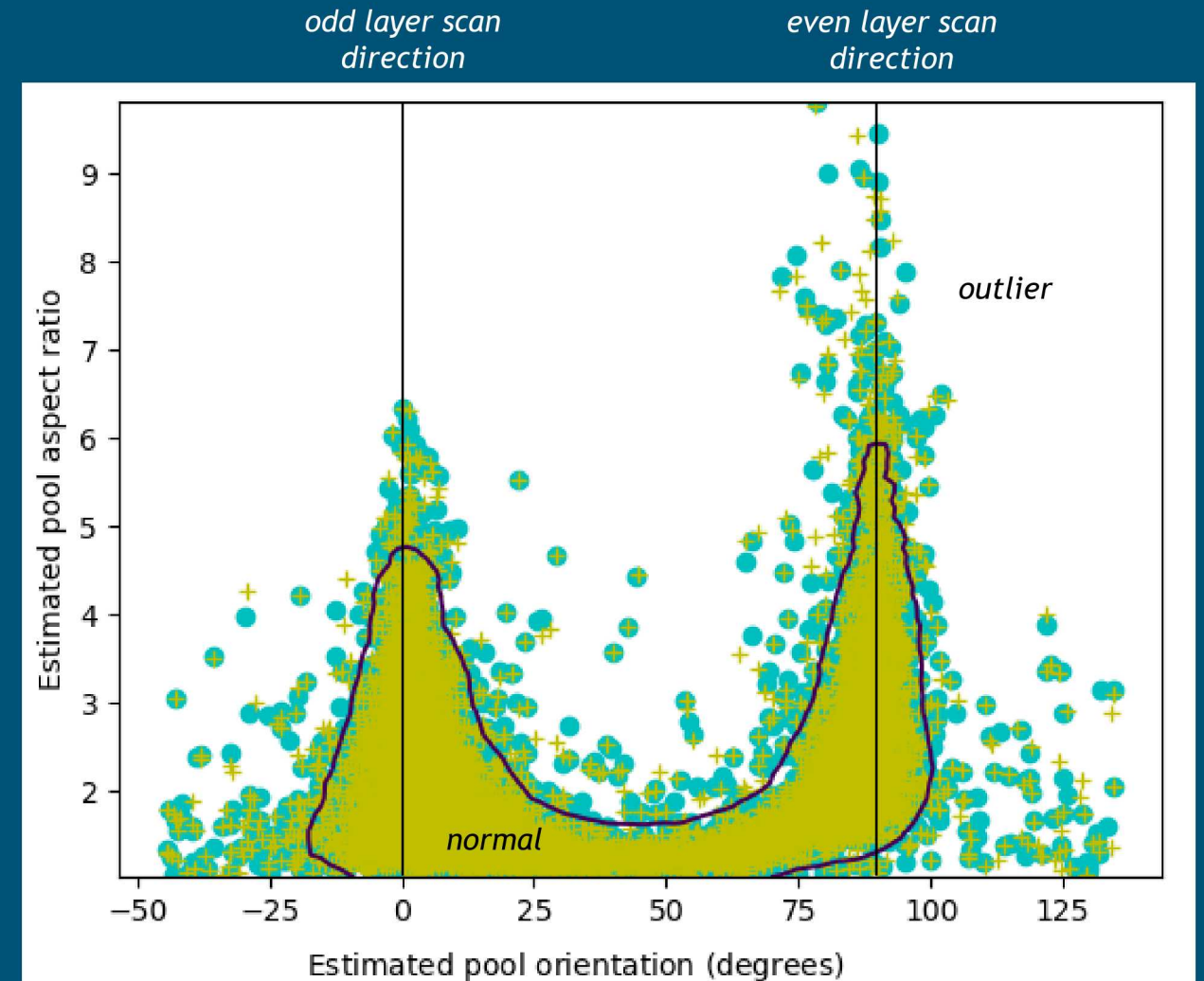
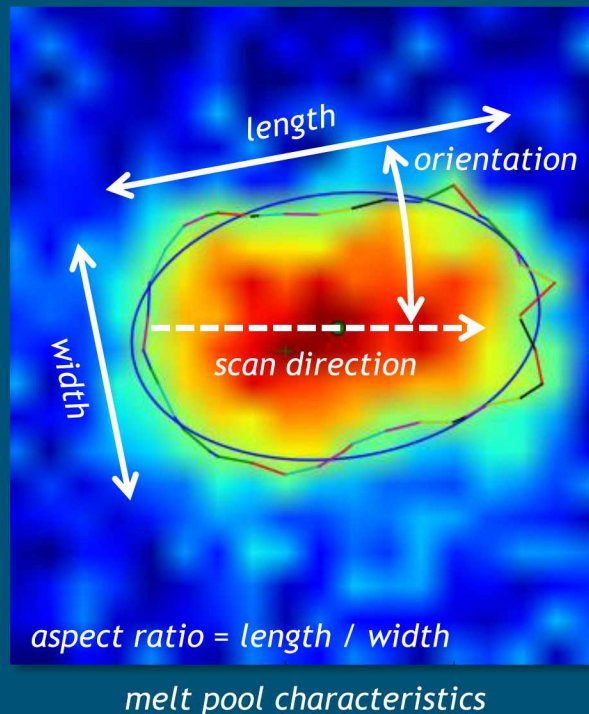


Calculate melt pool characteristics (length, width, orientation, ...)

- aspect ratio vs. orientation determined valuable

Classify normal & outlier melt pools by neighbors

- outliers had <50 neighbors within a search region
- 457 outliers identified





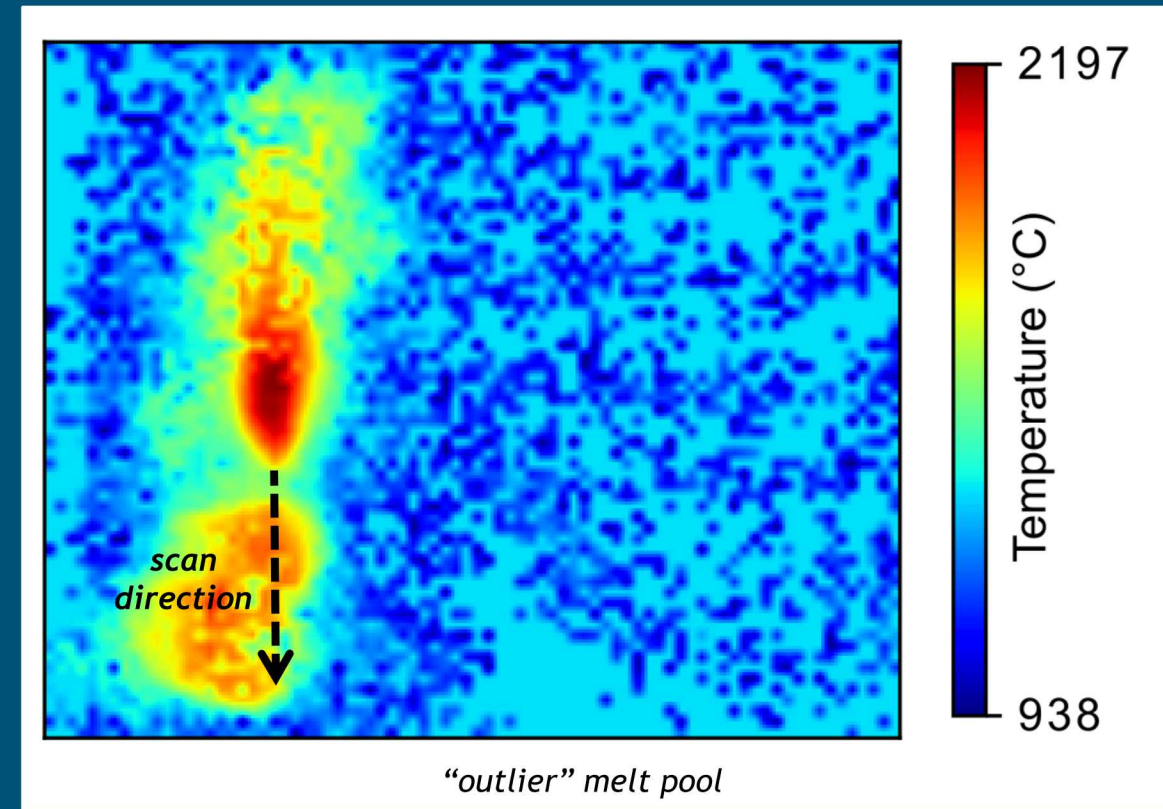
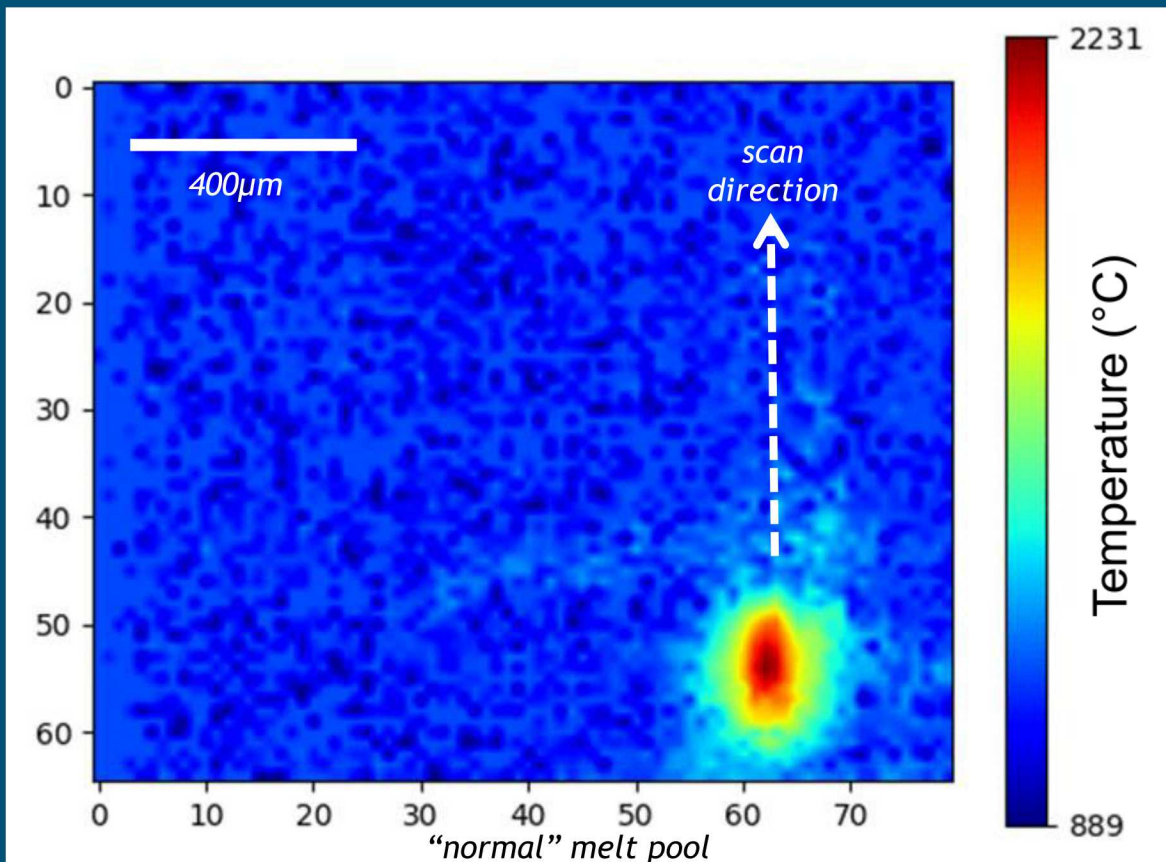
# Melt Pool Outliers

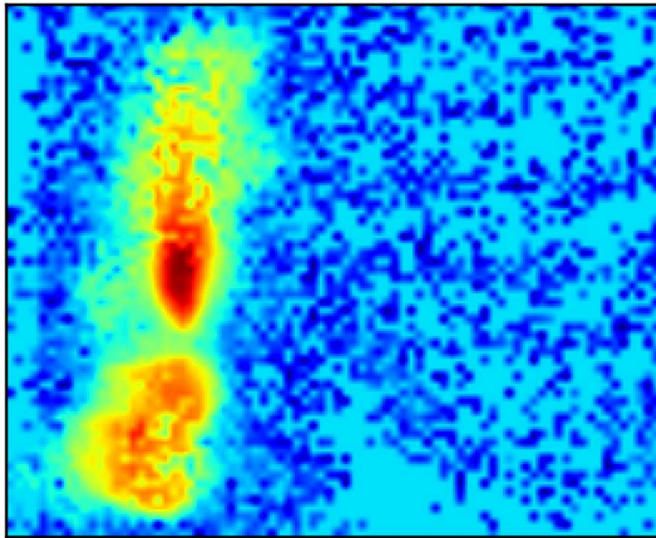


Normal pools are elliptical, symmetrical & aligned to scan direction

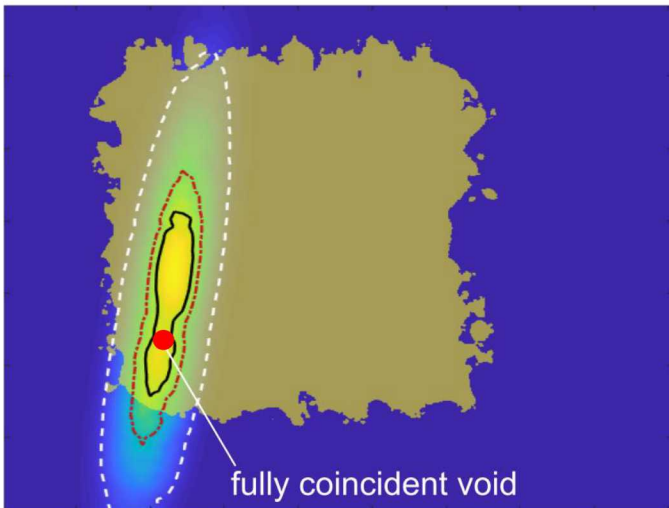
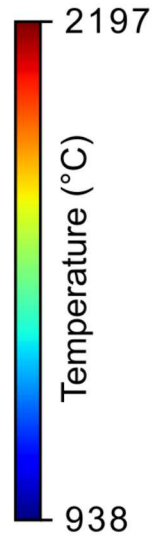
Outliers are not

- orientation angle outliers most likely signify spatter
- aspect ratio outliers may signify overheating / keyholing





*"outlier" melt pool*



*overlay of melt pool contour with pore location*



*All voids (iso view)*



*Correlated voids*

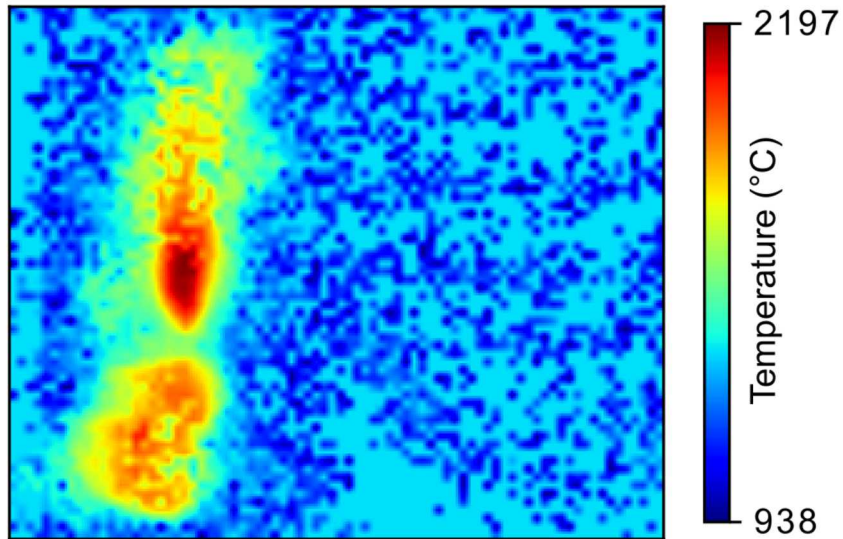
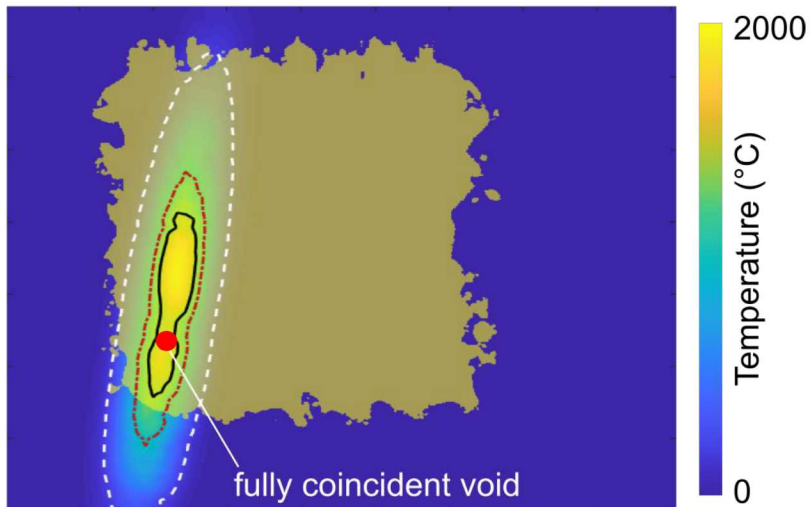
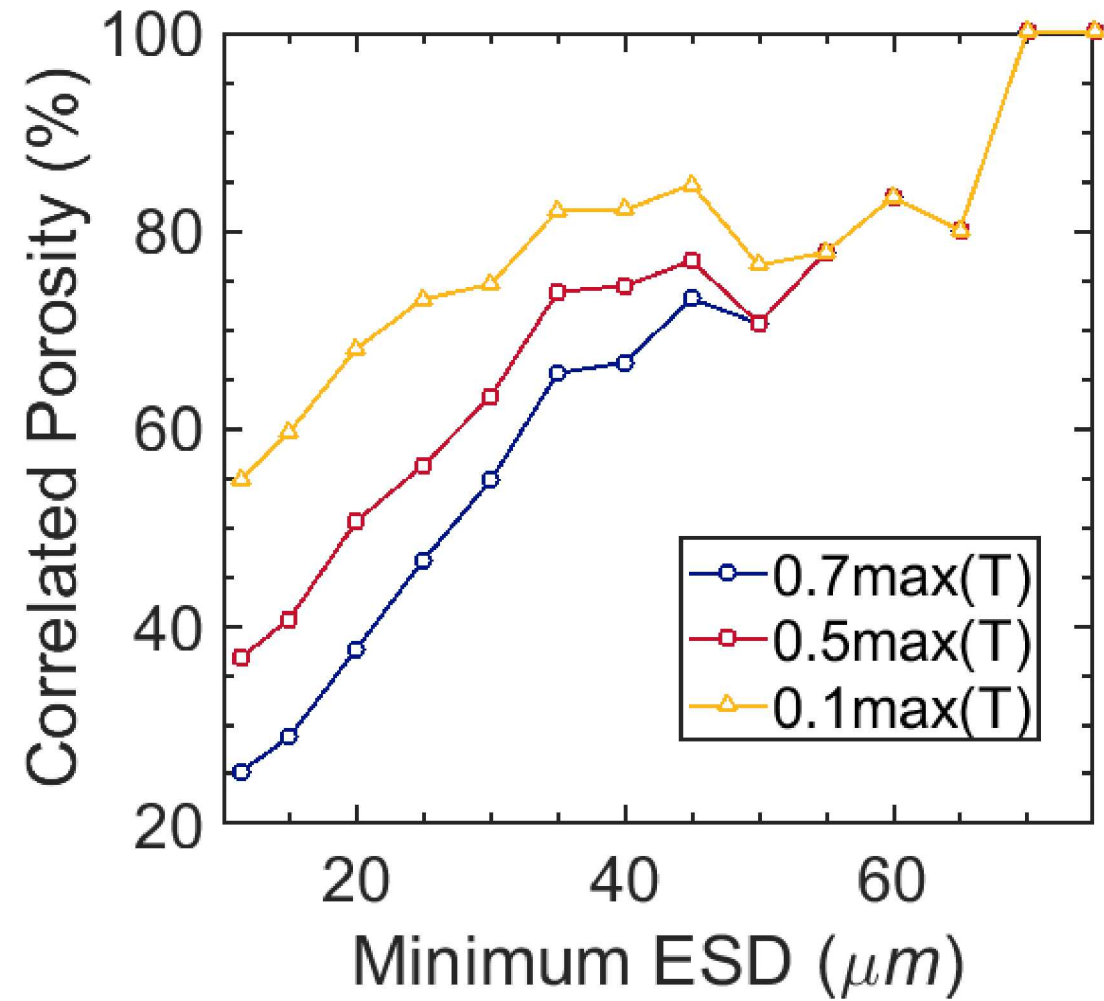


*Non-correlated voids*



Correlation established if pore(s) is located within search region surrounding outlier melt pool



*"outlier" melt pool**fully coincident void**overlay of melt pool contour with pore location*

Correlation success increases with increasing pore size

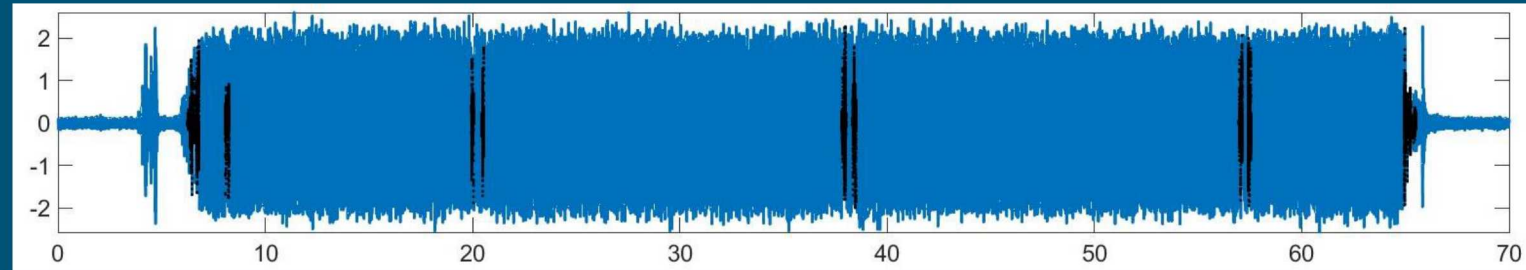
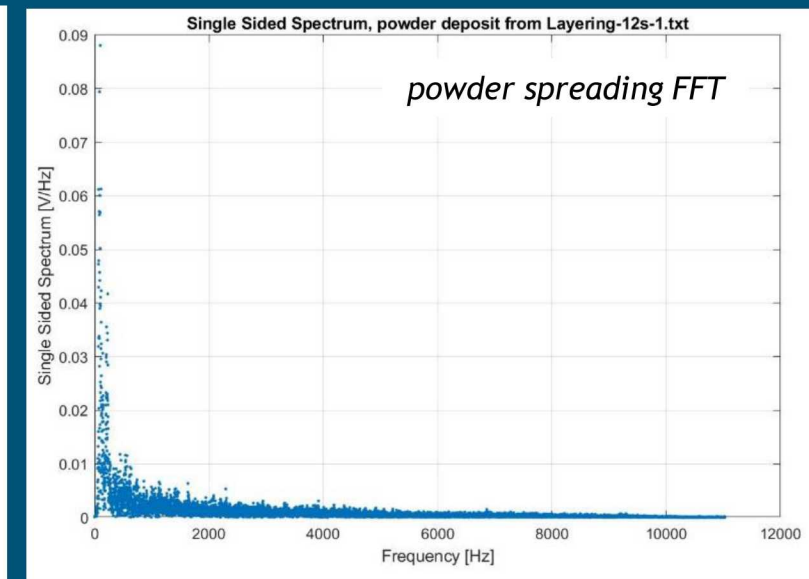
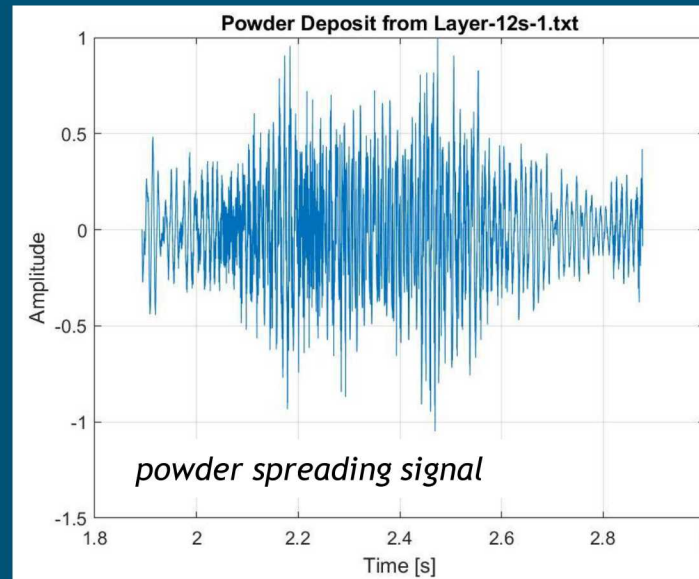
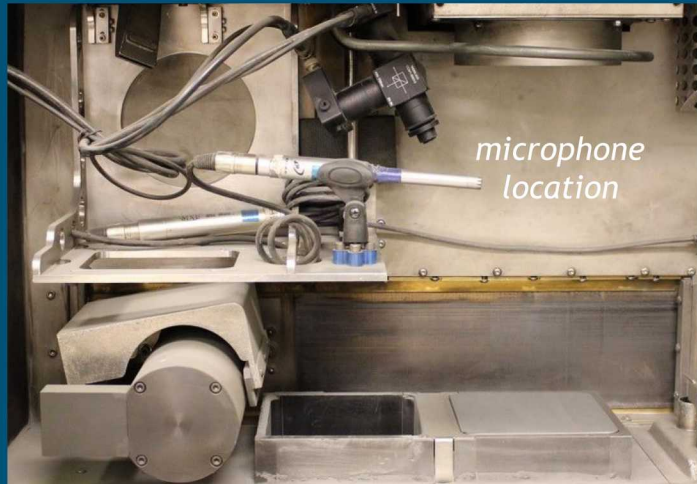
# Acoustics

Simple, inexpensive sensors

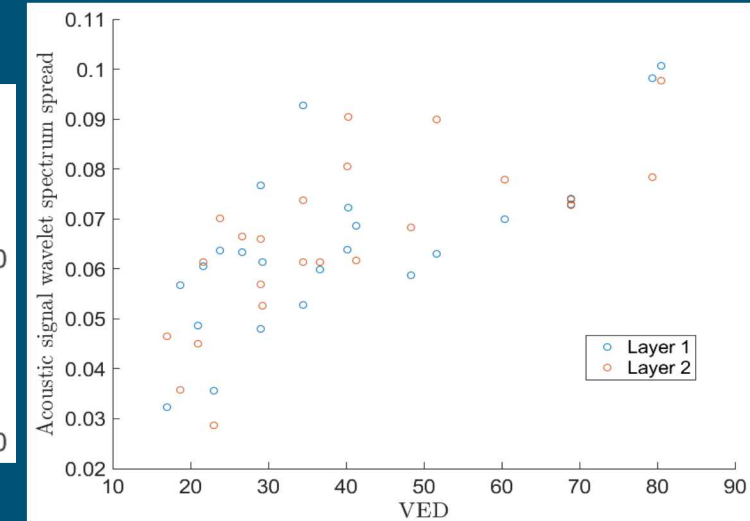
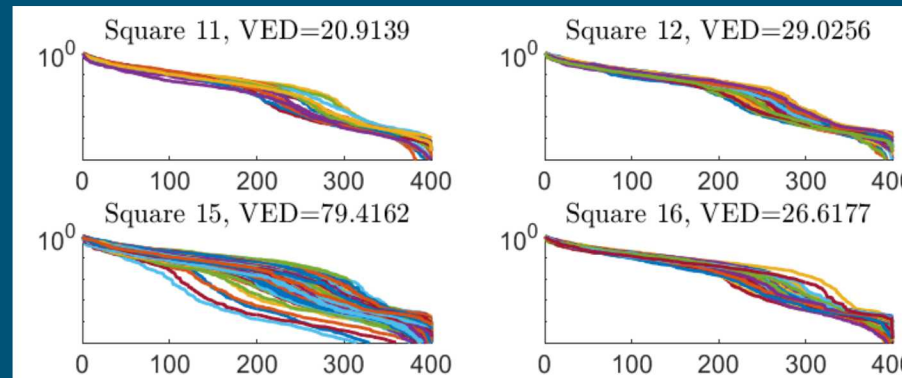
Interrogation potential

- welding prior art
- melt pool, powder spreading, part damage, environment disturbances

Challenging signal deconvolution



wavelet analysis of varying laser power & velocity







Questions?

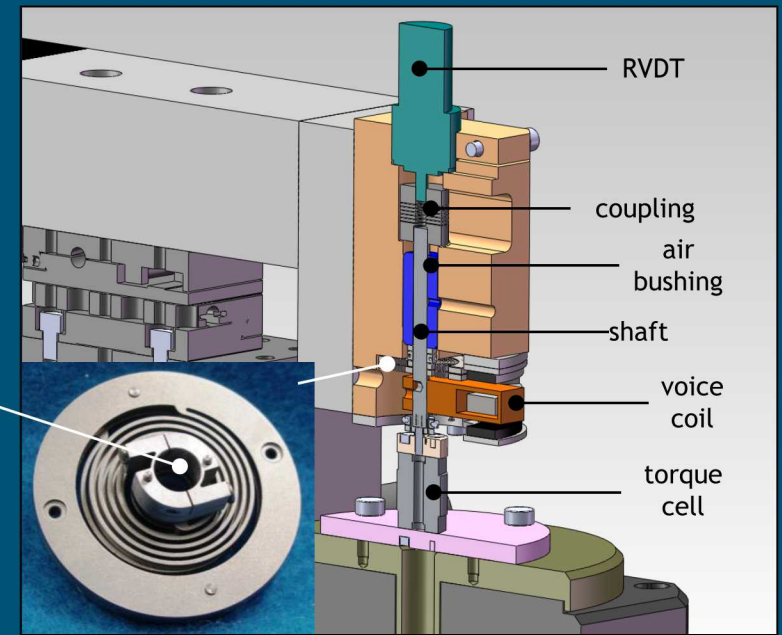
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Bradley Jared, PhD

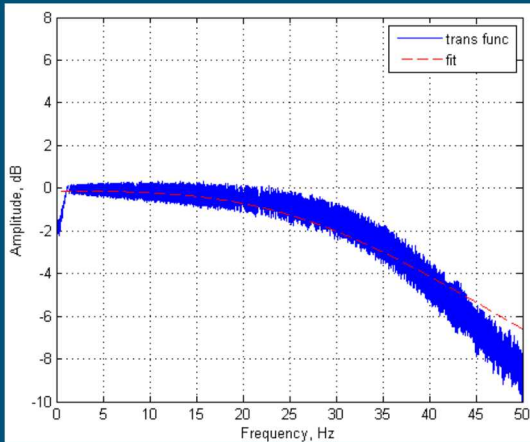
# Dynamic Meso-Scale Torsion Spring Metrology



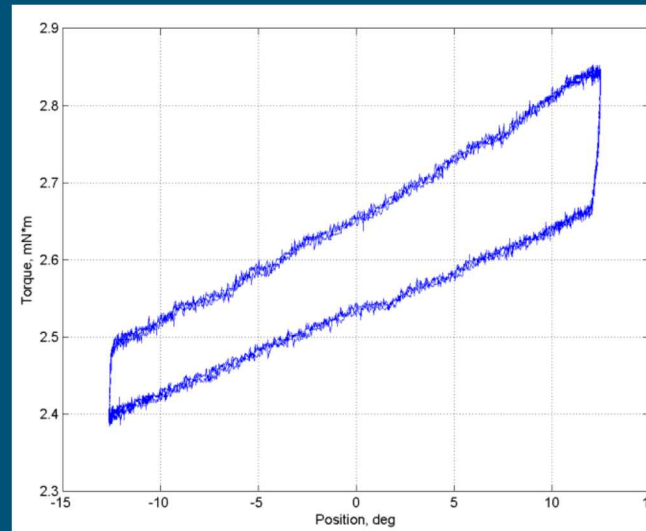
A representative meso-scale torsion spring. The wire diameter is 0.175mm, the spring length is 3.8mm, the coil diameter is 2.5mm, and the wire material is Elgiloy.



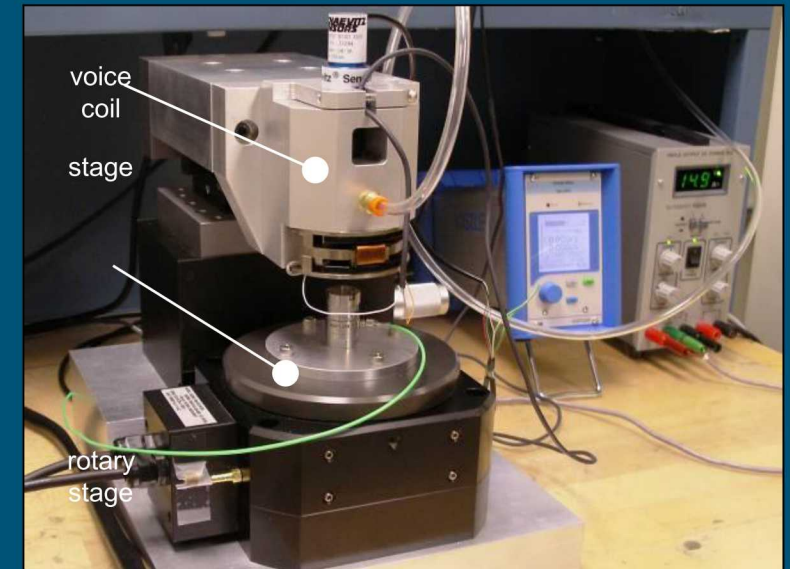
A cross-sectional view identifying critical elements of the actuation mechanism.



Voice coil actuator PID closed loop transfer function.



Torsion spring data from a 1Hz,  $\pm 12.5^\circ$  sine wave with a  $82.5^\circ$  preload.

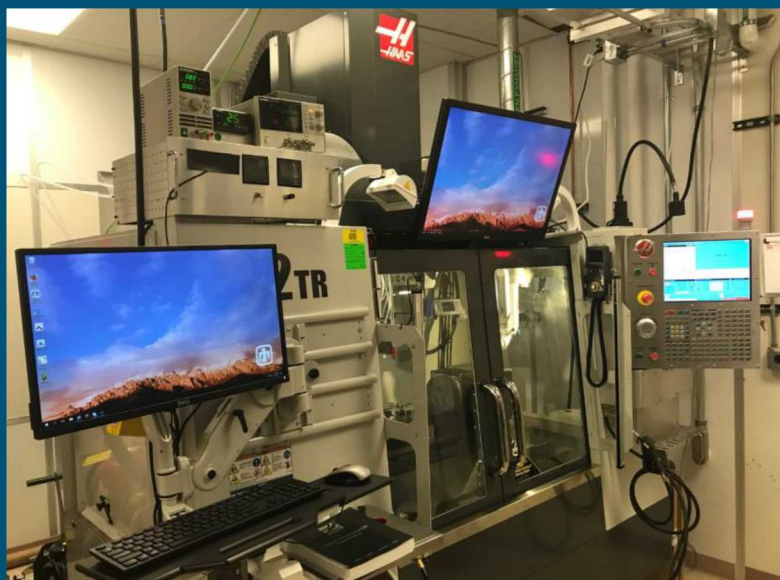


The prototype system for the measurement of meso-scale torsion springs.



## New capability for SS, Ti & Cu

- plasma arc welding torch (PAW)
  - MIG head ordered
- retro-fit onto HAAS VF2
- hybrid additive / subtractive capable
- good microstructures observed
- addressing growing needs for ES&H & larger parts
  - 300x300x150mm build volume
  - 2-6mm weld bead size



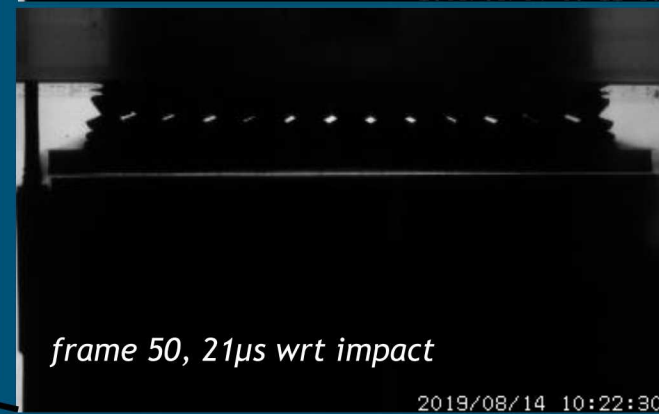
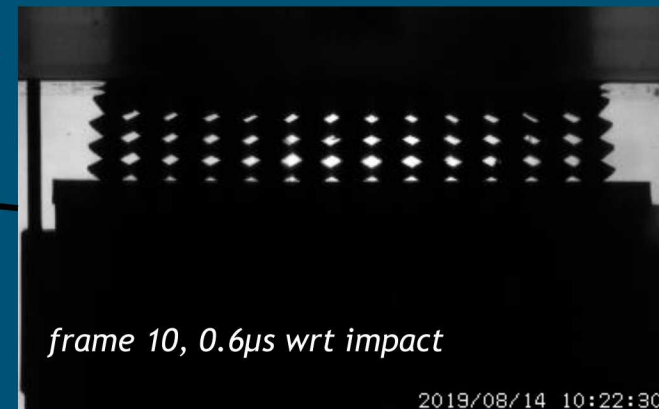
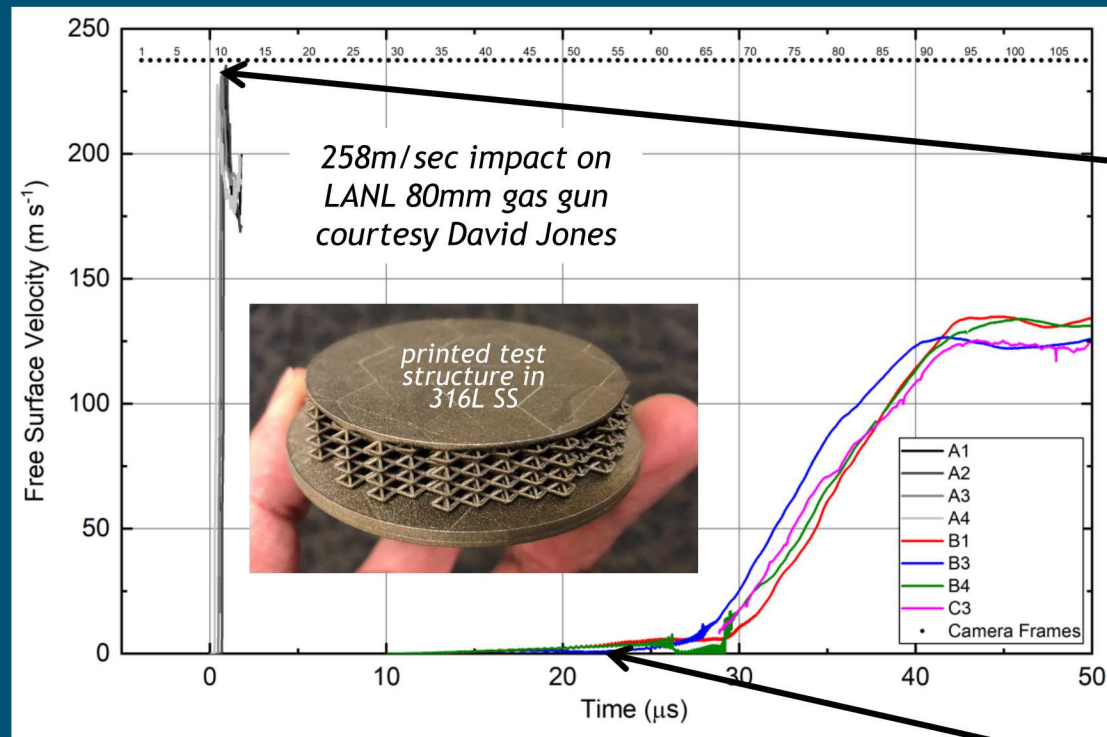
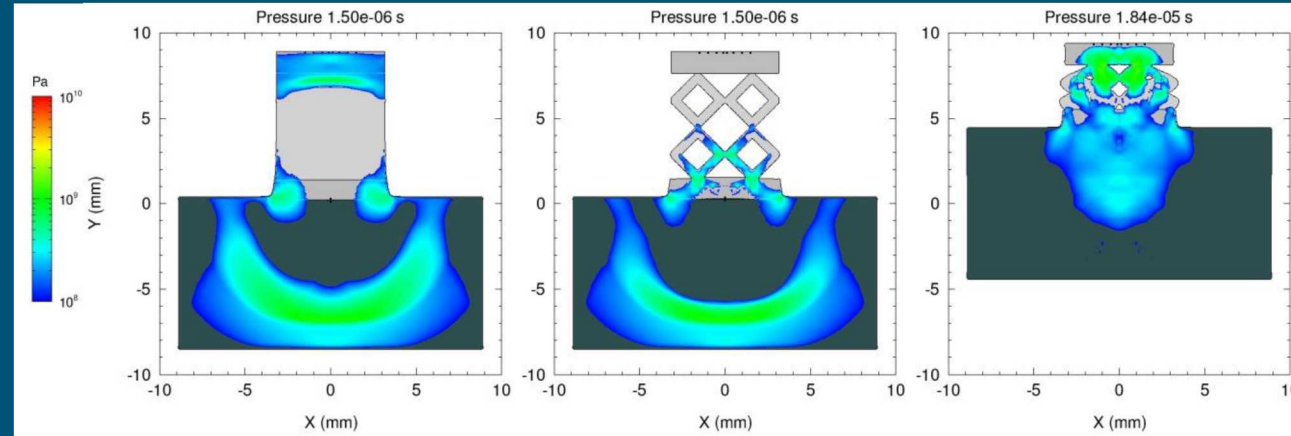
HAAS VF2 mill-turn



Ti-6Al-4V printing



baseline: shock arrives  $1.5\mu\text{s}$  after impact (left)  
 lattice: no shock @  $1.5\mu\text{s}$  (center), shock arrives @  $18.4\mu\text{s}$  (right)







“Complexity is free” has been an oft-cited refrain during the recent renaissance of manufacturing, typically referencing additive manufacturing (AM). To be sure, however, advanced manufacturing techniques make complexity available as new materials, processes and part topologies are realized enabling performance gains inaccessible through traditional means. Such gains will be discussed in the context of ultra-fast laser processing, precision machining and laser-powder bed fusion (L-PBF). Application examples include meso-scale springs, freeform optics, topology optimized structures and hierarchical materials for impact mitigation.

While complexity may be available and desired in component material and geometry, complexity introduces costs that are inevitably accrued during development, qualification and inspection activities. Immature industries and processes routinely address this complexity using heuristics, human expertise, administrative controls, experimental validation and/or ex-situ inspection. Such approaches can be expensive and prohibitive in terms of time and resource commitments. Instead, the development of physics-based engineered controls and automation methodologies are preferred to satisfy requirements by optimizing process performance, by minimizing output uncertainties, and by increasing product throughput.

L-PBF is one of the most popular forms of metal AM as it affords designers the ability to fabricate complex structures such as lattices and topologically optimized structures. The adoption of L-PBF in high consequence applications, however, has been relatively slow due to process complexities and uncertainties in resultant material properties. Research on L-PBF will be presented to address these process complexities and performance uncertainties. Initial material behavior mimicked an open loop state as properties varied widely and proved difficult to anticipate. Work has since quantified important process-structure-property relationships for powder feedstocks, laser settings and machine components. Machine improvements and metrology procedures have also been developed, producing an increase in process consistency and material confidence. Continued research is examining in-situ melt pool dynamics and exploring advanced data analytic techniques (ex. machine learning) to correlate melt pool outliers to process defects. A desired future state is real-time control of L-PBF melt pools to continue process improvements and to reduce defects. A further goal is to inform and support an accelerated qualification paradigm for high consequence parts whereby development and production cycles are reduced dramatically over current timescales and resource allocations.