

New Insights into Alloy Design for Tribological Applications



PRESENTED BY

Dr. Nicolas Argibay



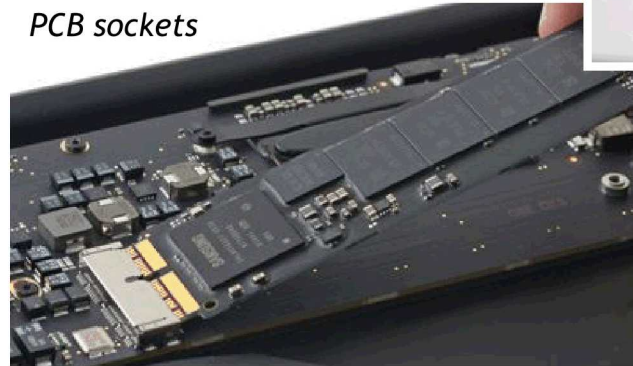
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Motivation for Metals Tribology Research

wind turbine slip-rings
(sensors and blade pitch motors)



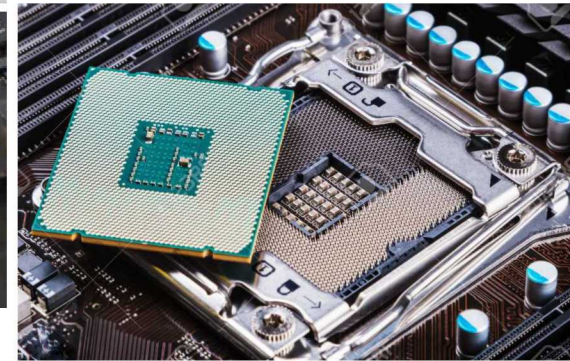
PCB sockets



cell phones



CPU sockets



**Estimated 150 Metric Tons (\$6.9B) of Au
used in Electrical Contacts per Year:**

*Refs: Gold Survey, Gold Fields Mineral Survey Ltd, 2011
Gold Bulletin 2010, Vol. 43-3, C. Hagelüken and C.W. Corti,
Gold Bulletin 1986, Vol. 19-3, T.D. Cooke*

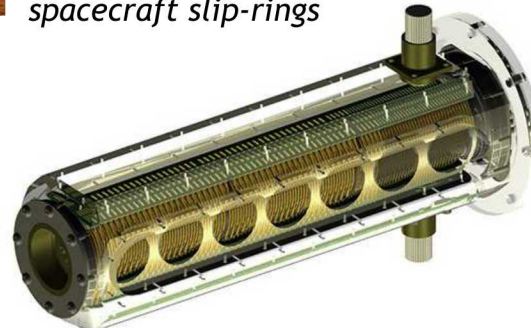
45 connectors



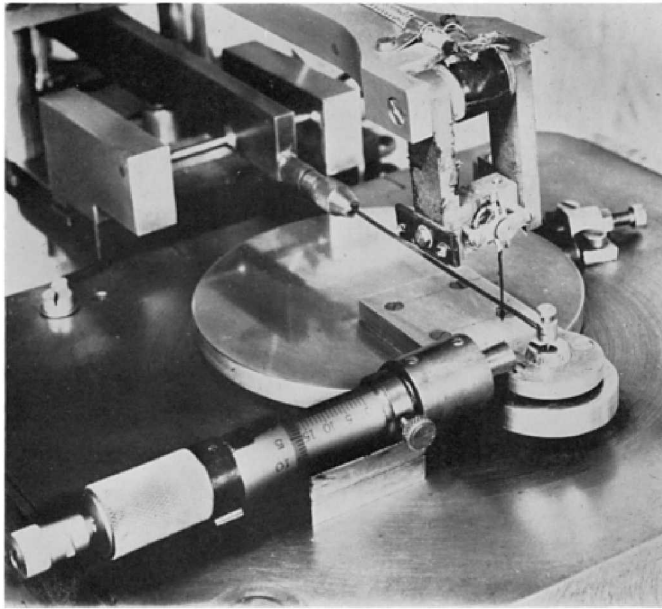
EV charging



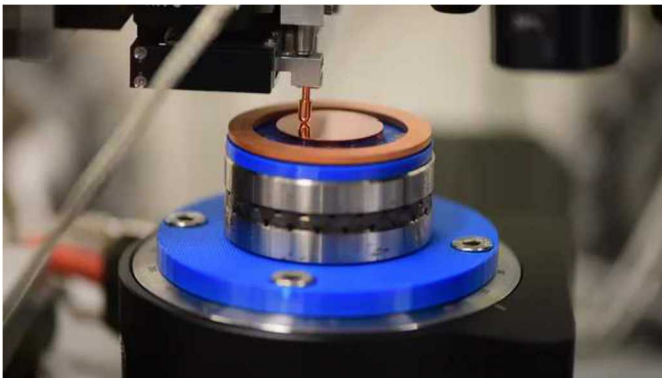
spacecraft slip-rings



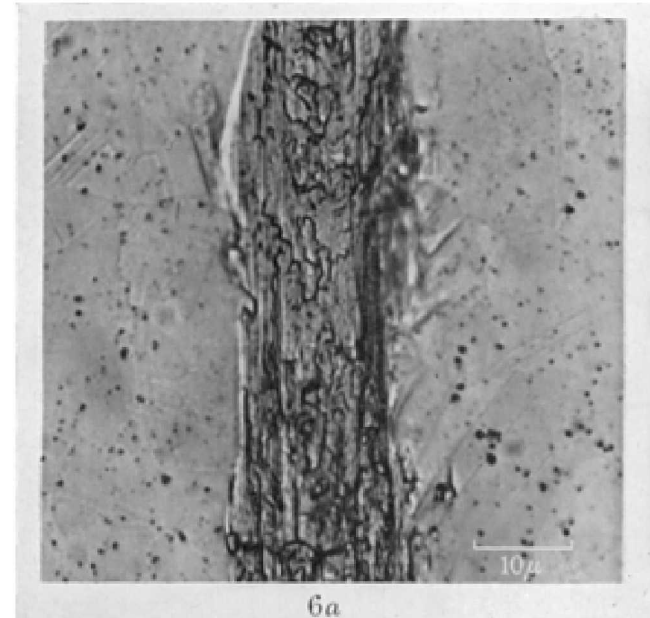
Metals Often Have Poor Tribological Properties



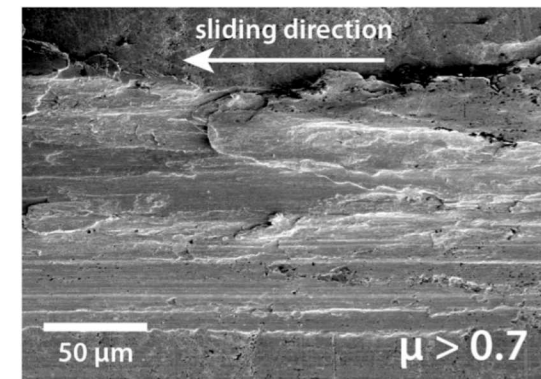
Whitehead, Proc. Royal Soc, 1950



Metals tend to show strong, rapid interfacial bonding (cold welding) with galling.

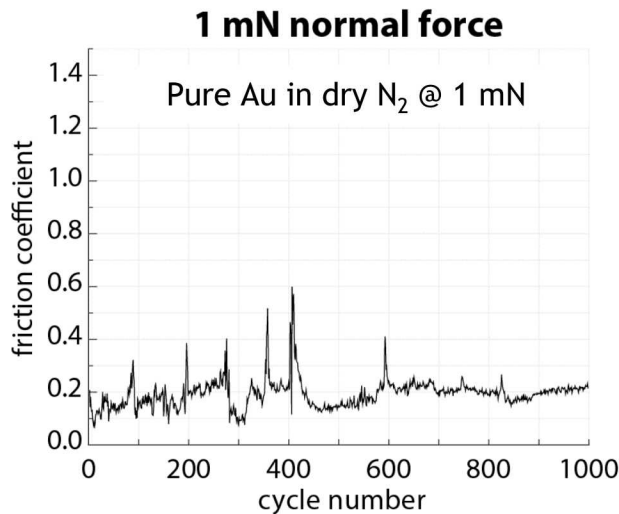


Cu in air @ 150 mN, $\mu > 1.6$



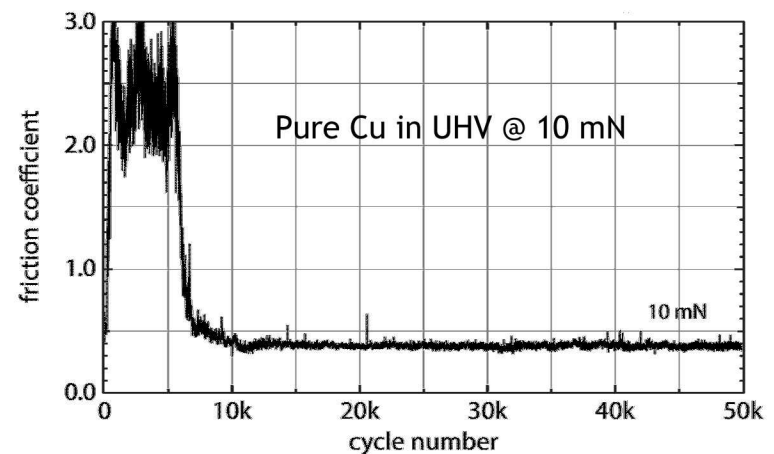
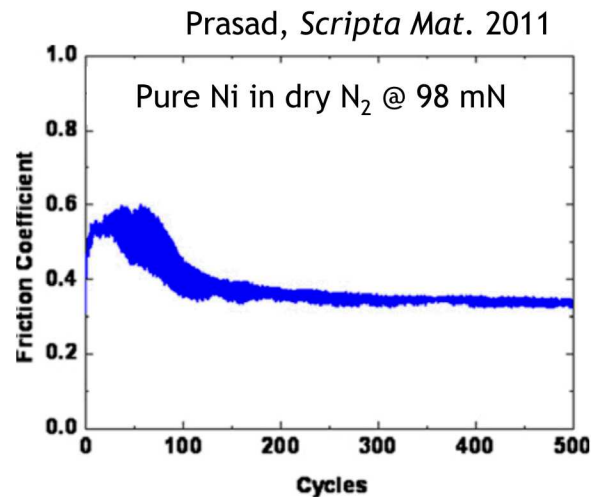
Au in dry N_2 @ 75 mN, $\mu > 0.7$

More recent results: Low Friction starting from Bare, Soft, Pure Metals



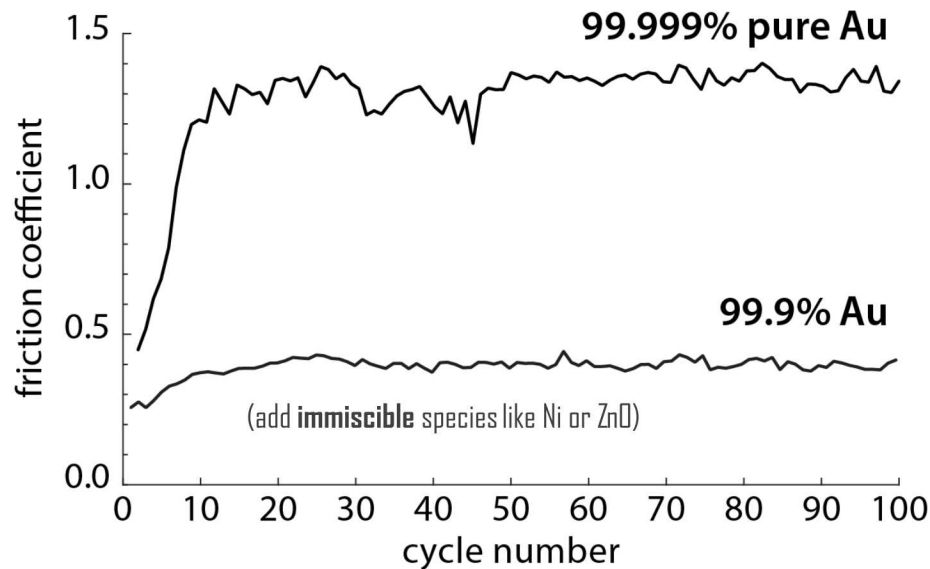
"When surfaces are cleaned in a good vacuum, the sliding friction... becomes vanishingly small."
- Bowden & Hughes, *Nature* 1938.

"It was found quite unexpectedly that with some metals, very low friction less than 0.10 was observed."
- Tamai, *J. Appl. Phys.* 1961



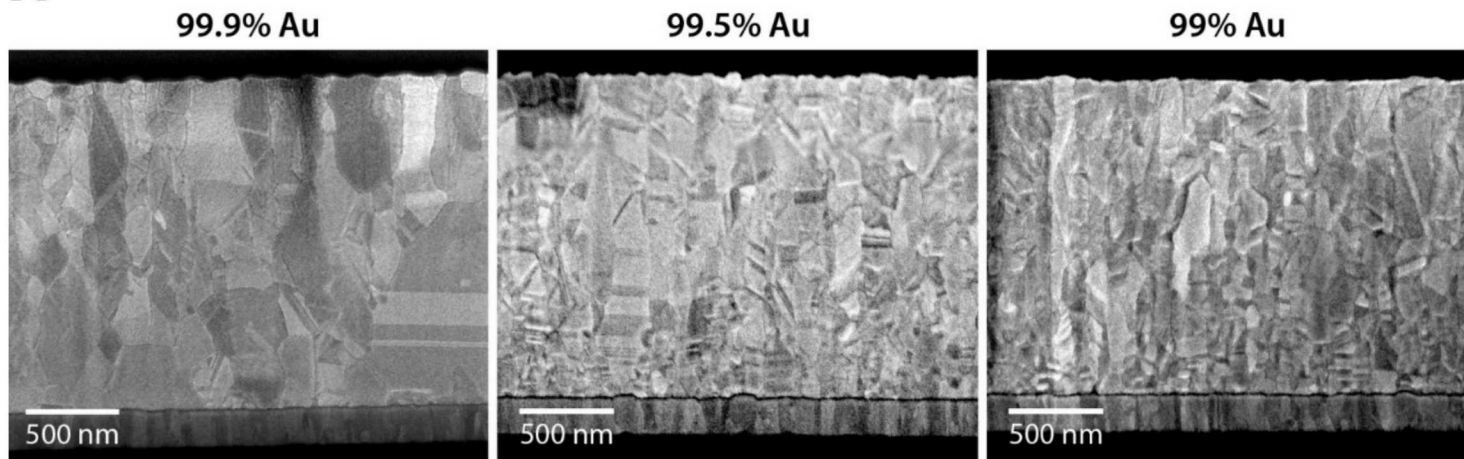
Low friction with pure metals is achievable.

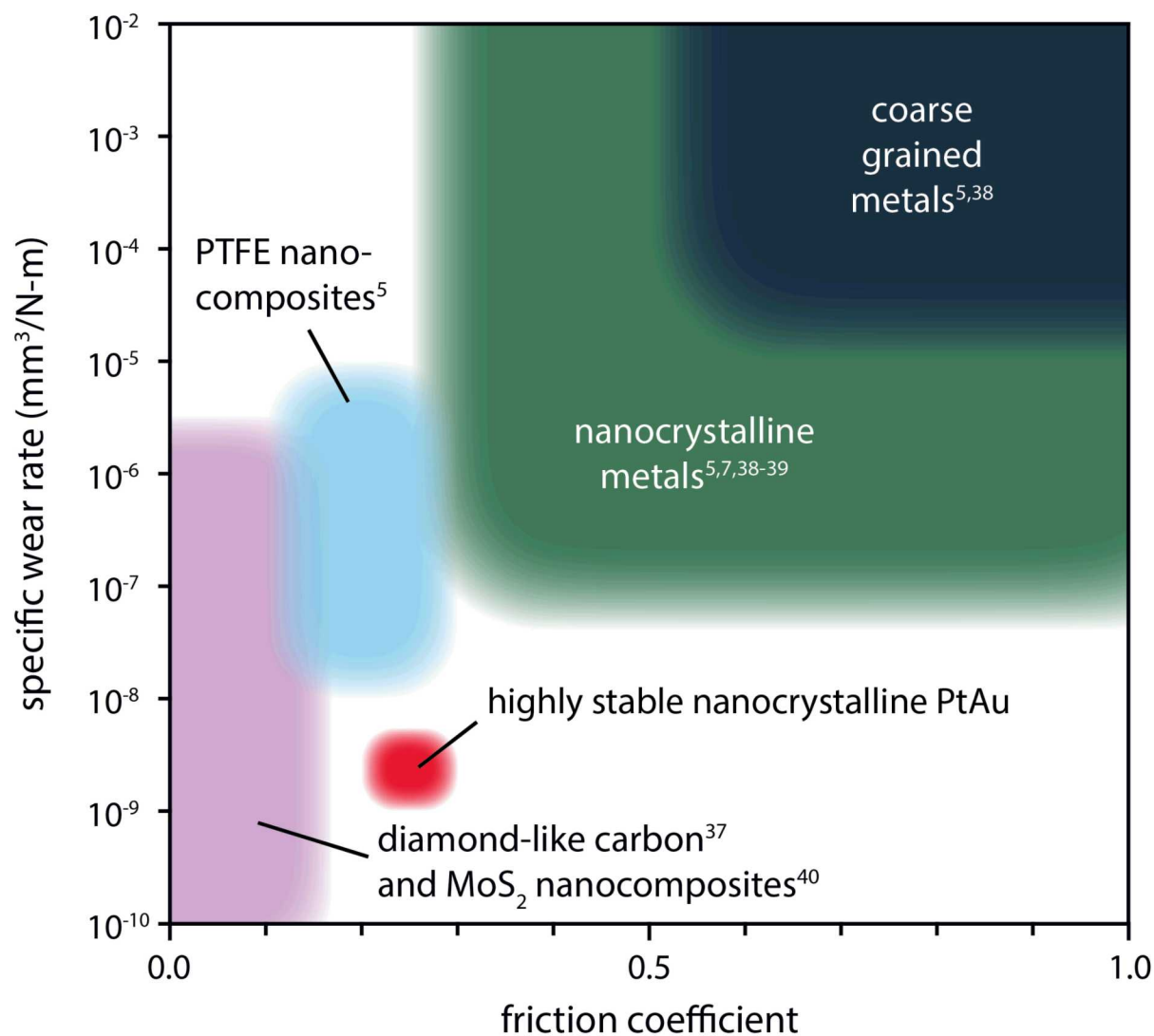
Alloying reduces friction coefficient:



Alloying improves friction & wear performance by reducing and stabilizing grain size

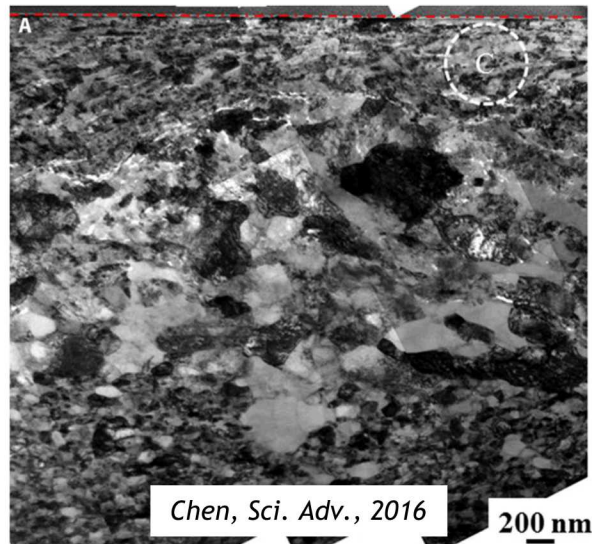
...by reducing grain size:



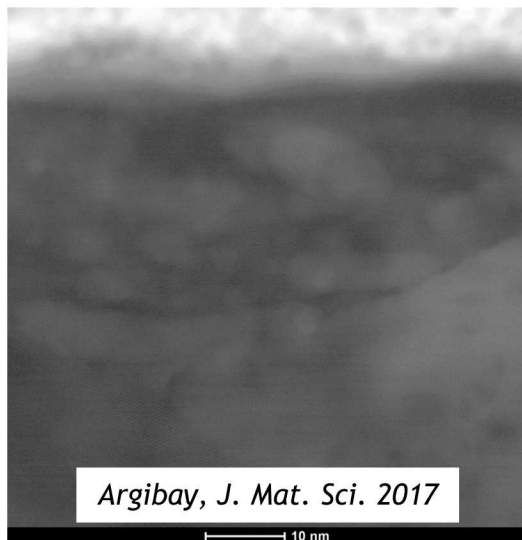


7 What do we know now?

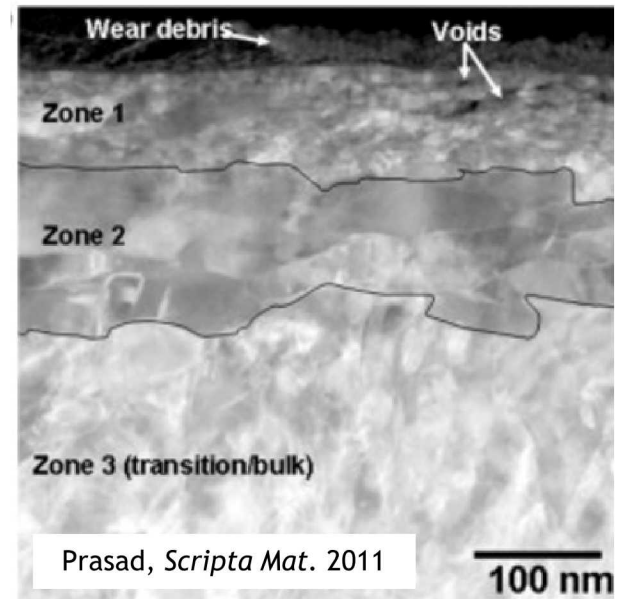
Pure Cu in air @ 50 N, $\mu \sim 0.37$



Pure Au in dry N₂ @ 1 mN, $\mu \sim 0.2$



Pure Ni in dry N₂ @ 98 mN, $\mu \sim 0.2$

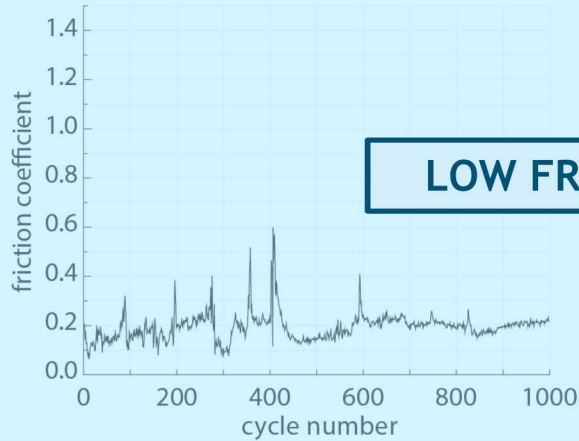


Low friction is associated with the formation of a highly surface localized UNC layer.

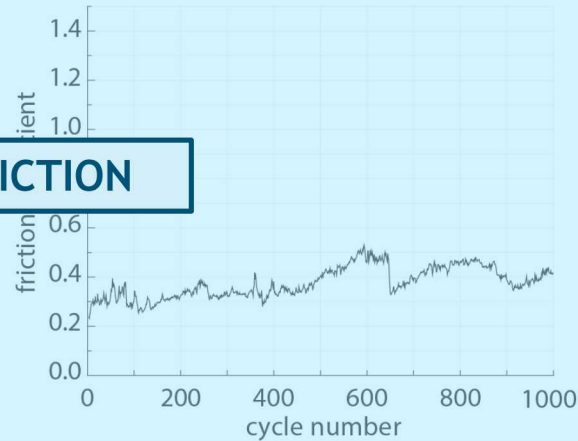
What do we know? Friction depends on applied load

(Pure Au in dry N₂)

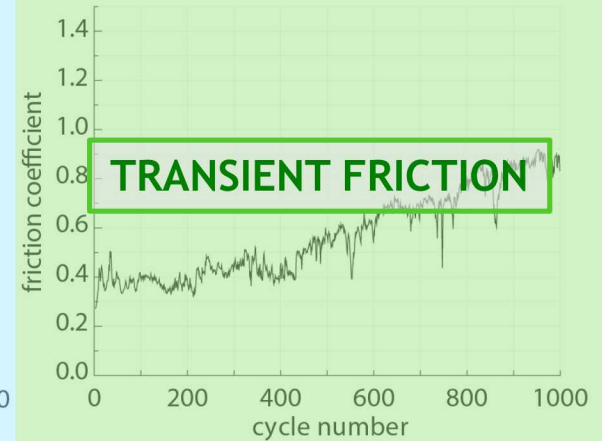
1 mN normal force



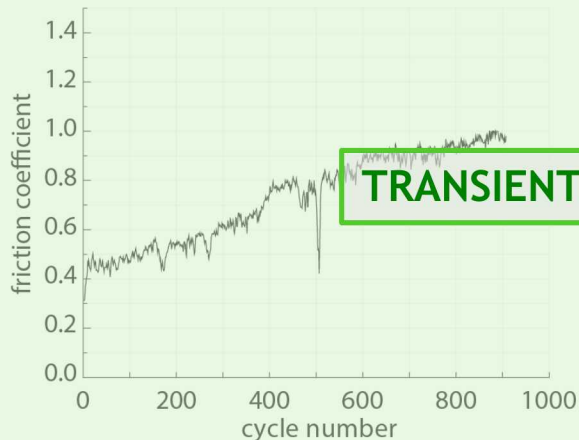
10 mN normal force



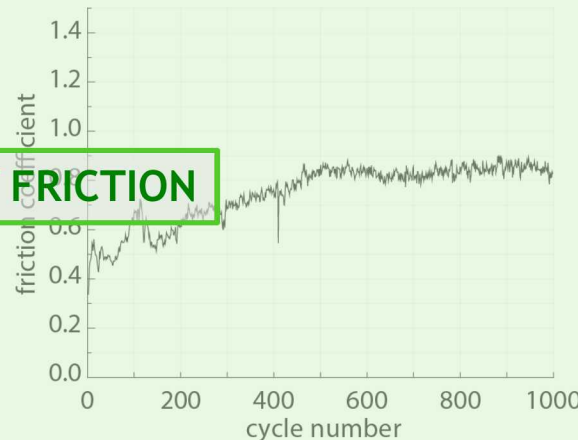
25 mN normal force



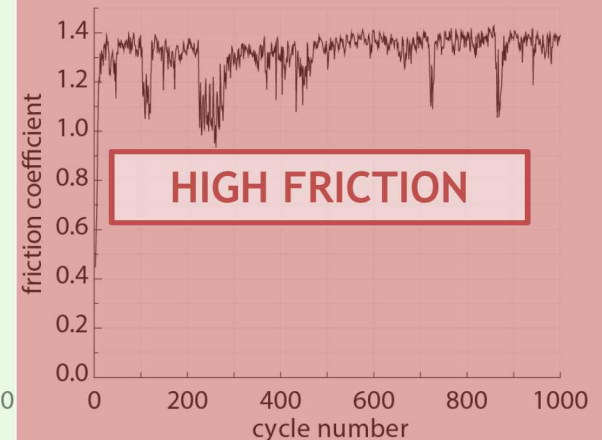
50 mN normal force



75 mN normal force



100 mN normal force

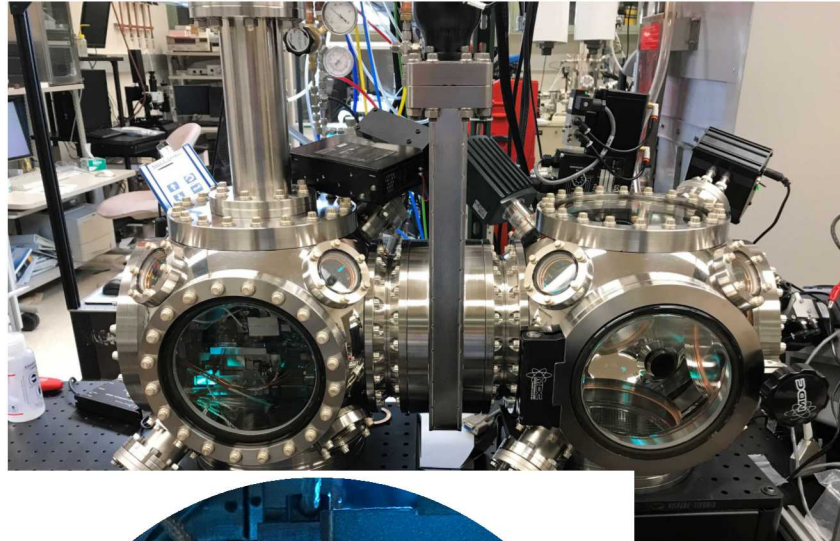
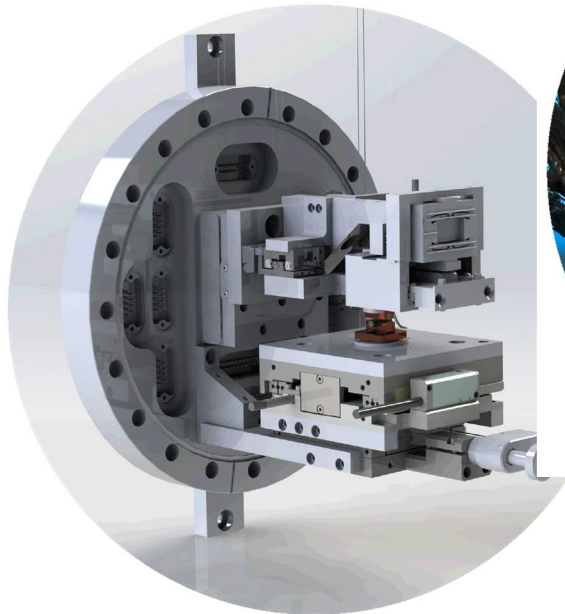


Three different friction regimes, with transitions.

9 What about adsorbates? Test in UHV

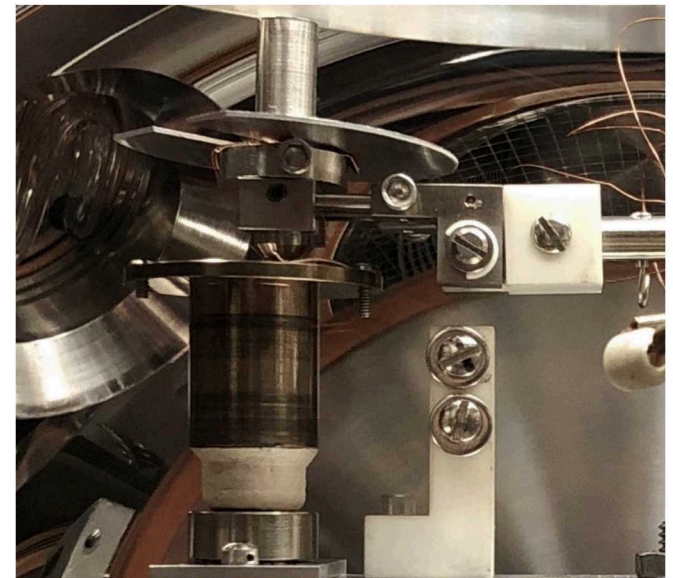
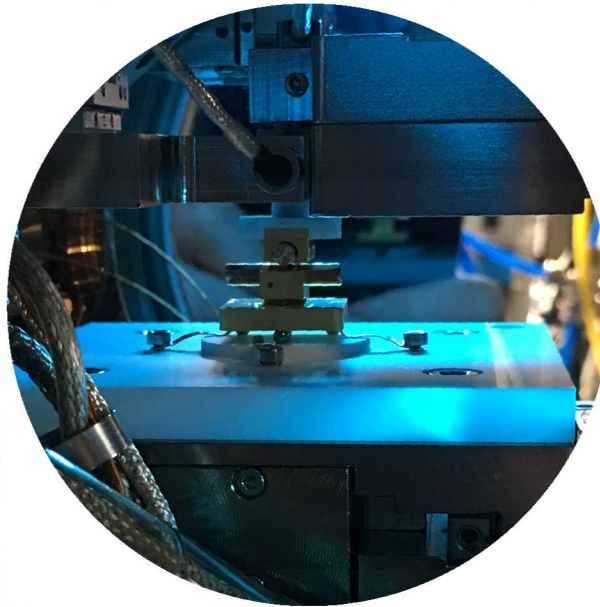
Left Module:

- Linear Reciprocator
- Load metering
- Cryo stage (4-800K)
- 10^0 to 10^{-9} torr
- 0.1 mN to 1 N
- 100 $\mu\text{m/s}$ - 100 mm/s
- 10 kHz acquisition
- capacitive displacement sensors & flexural cantilevers



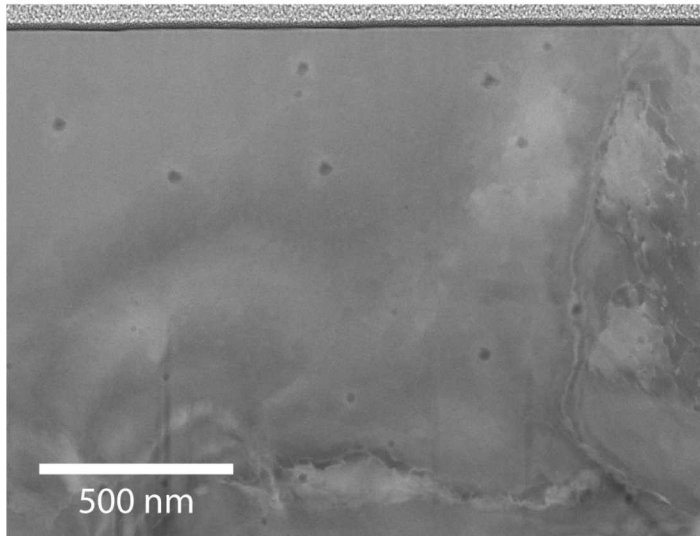
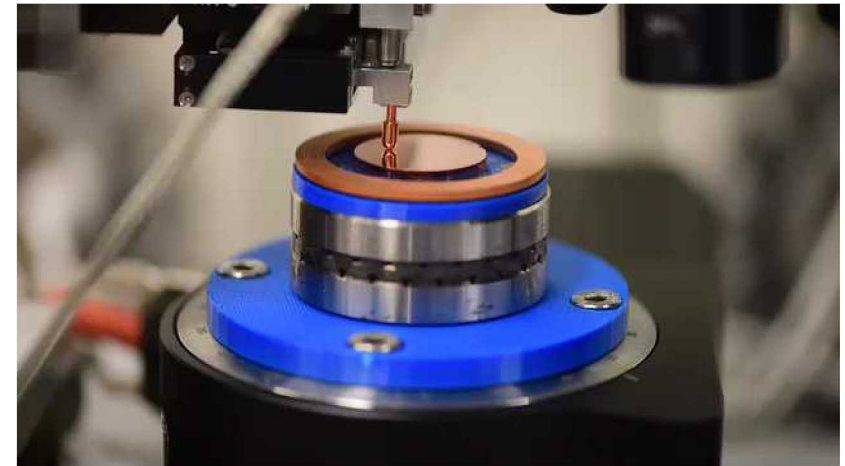
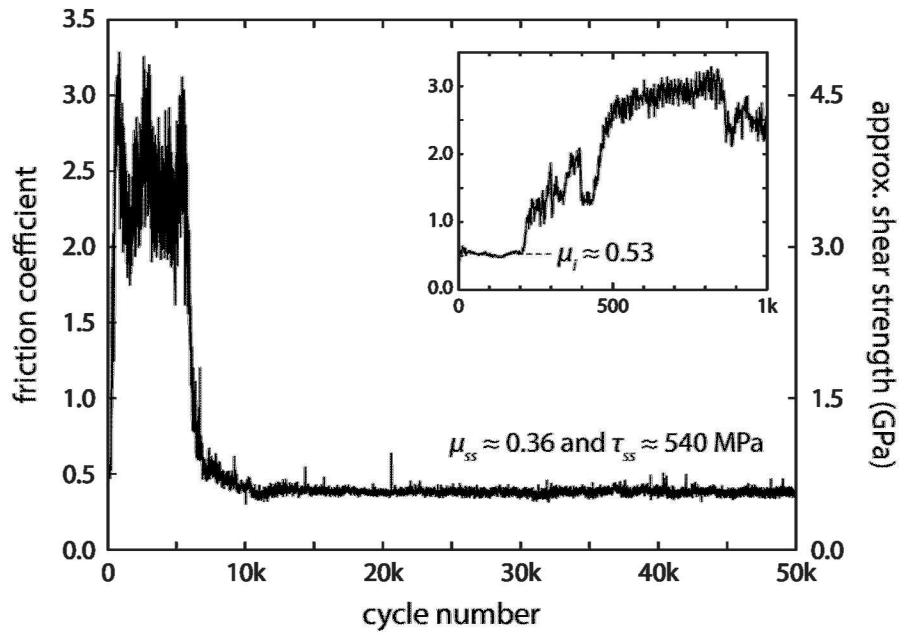
Right Module:

- Rotary Module
- Dead weight
- 5 mN to 10 N
- 100 Hz acquisition
- 10^0 to 10^{-9} torr
- Strain gage sensor

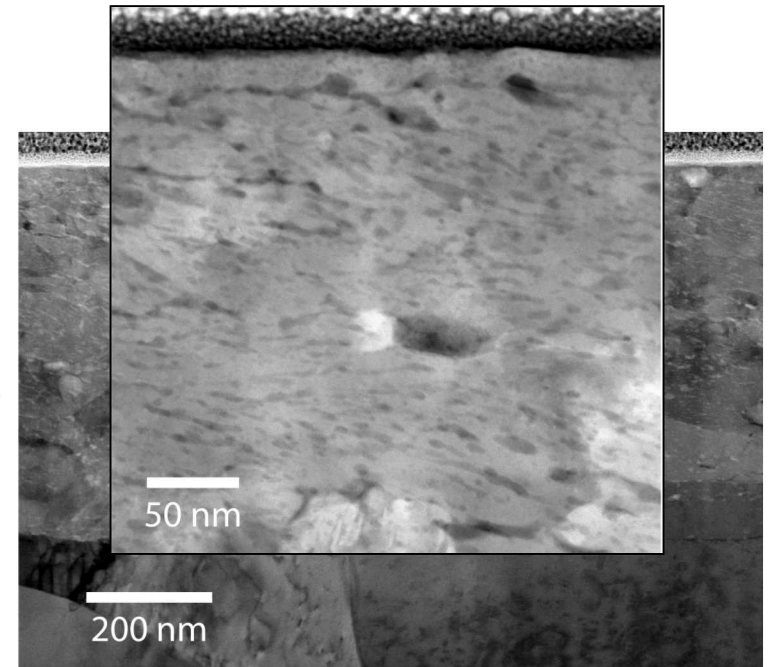


*crossed-cylinders
and sphere-on-flat
configurations*

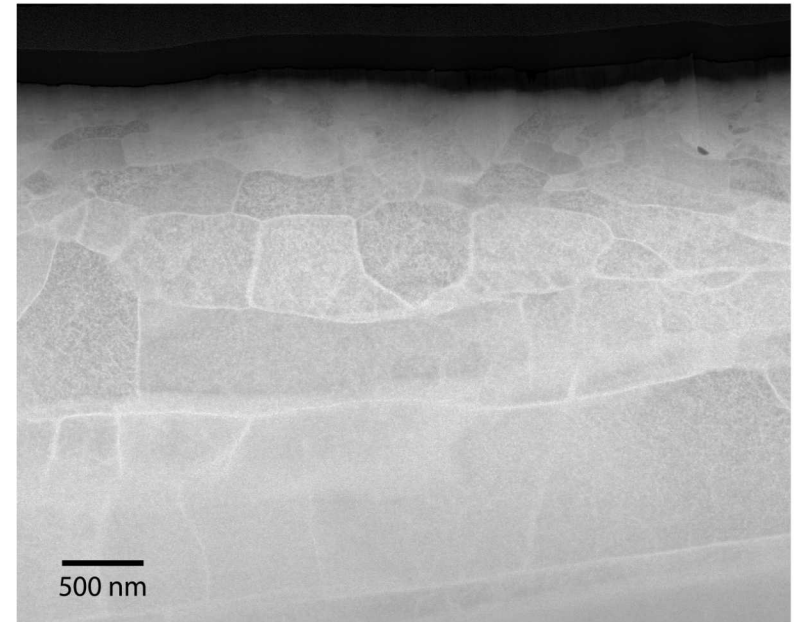
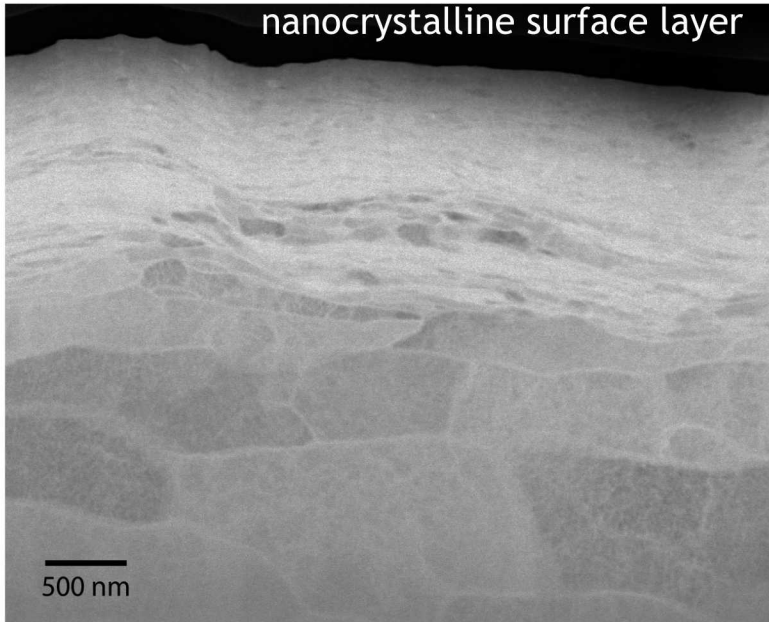
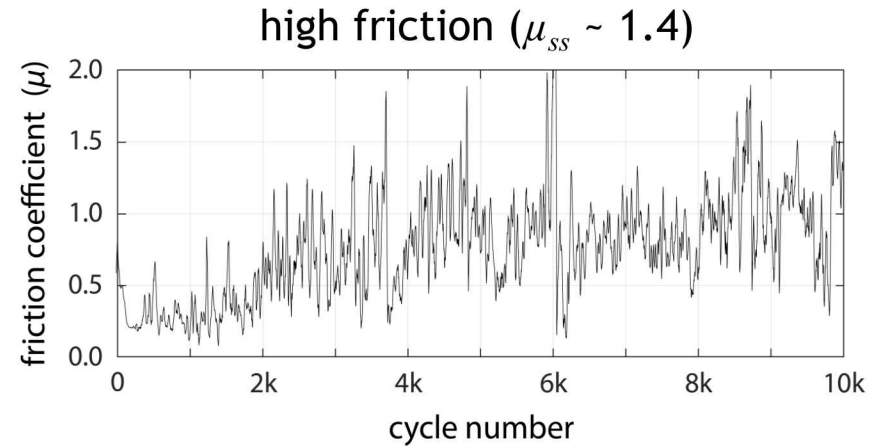
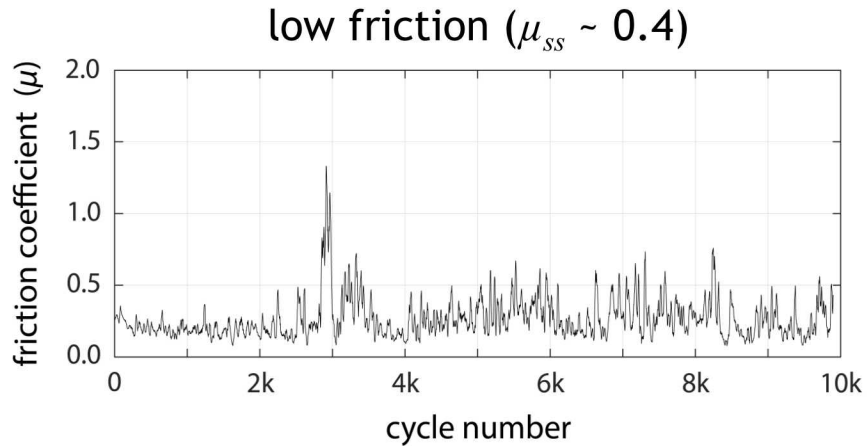
Low friction of FCC metals (pure Cu) in ultra-high vacuum



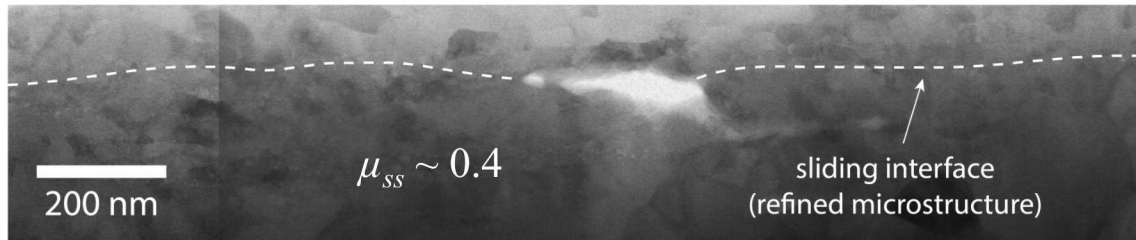
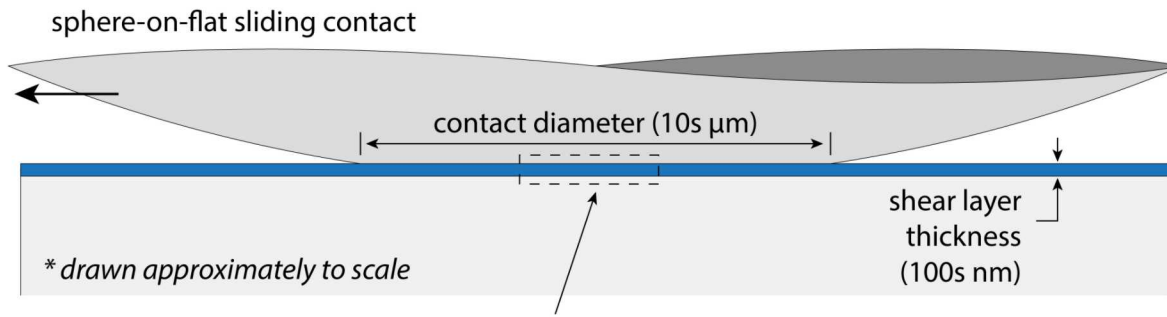
refinement



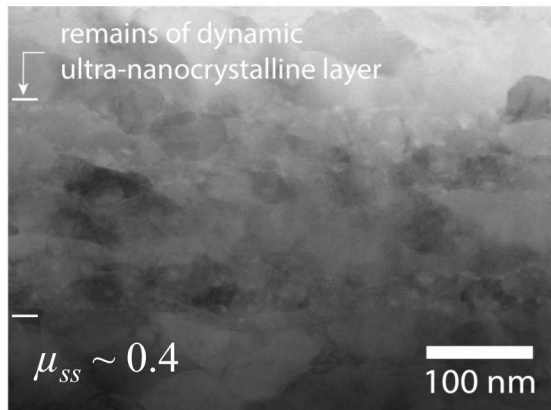
Low friction of BCC metals (pure Ta) in ultra-high vacuum



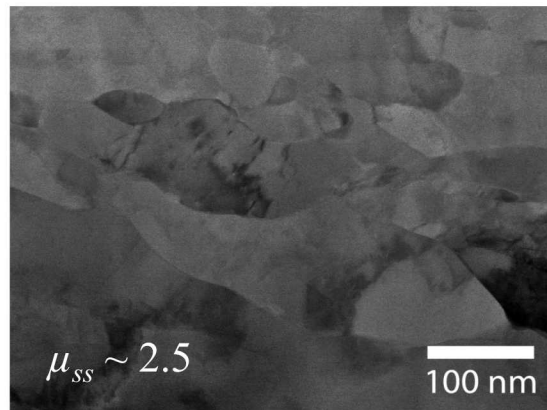
A cross-sectional view of metal sliding contacts



low friction and ultra-fine grains

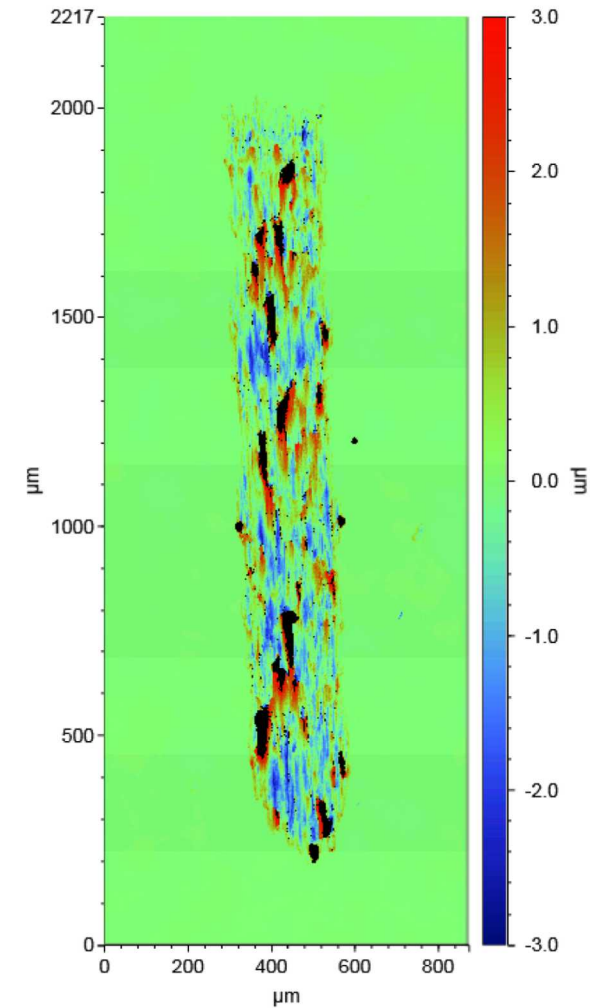


high friction and coarser grains

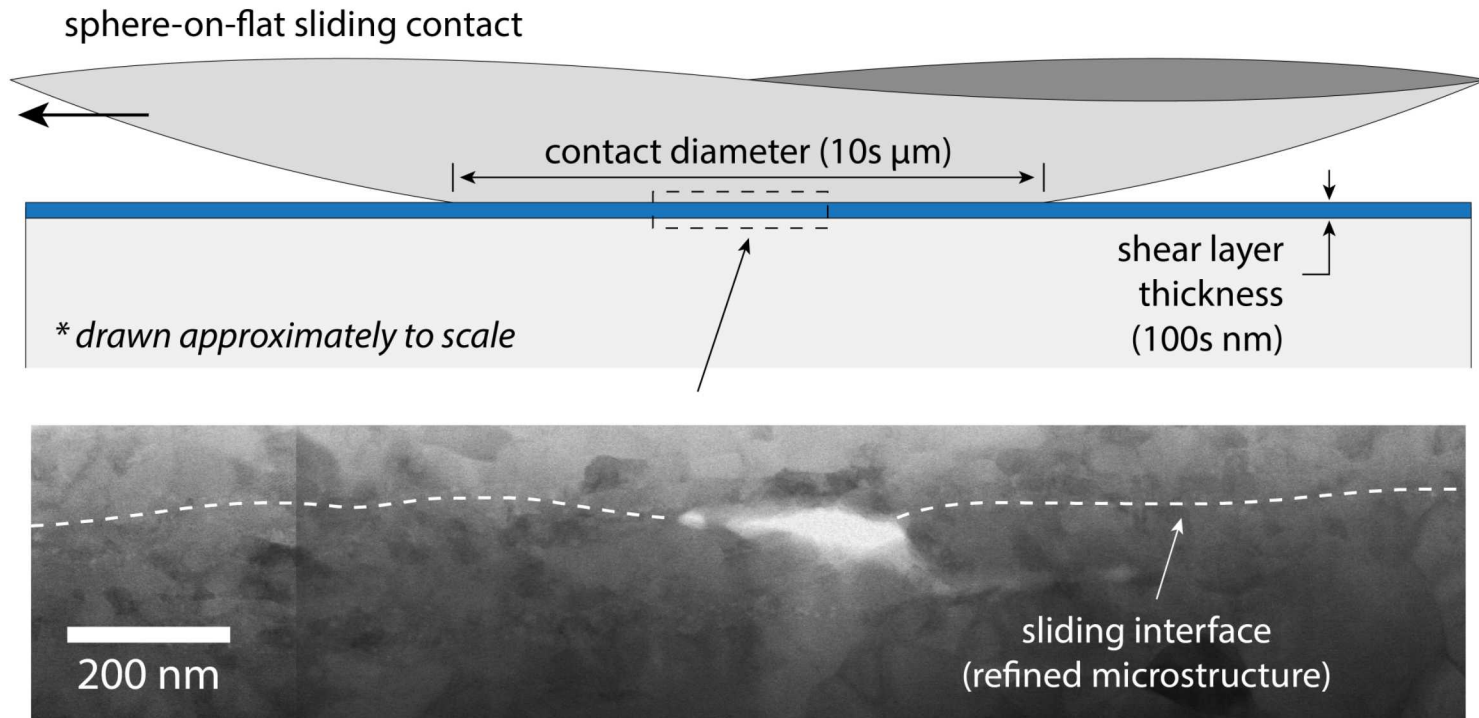


(TEM of self-mated pure Cu in dry N_2)

wear track topography map:



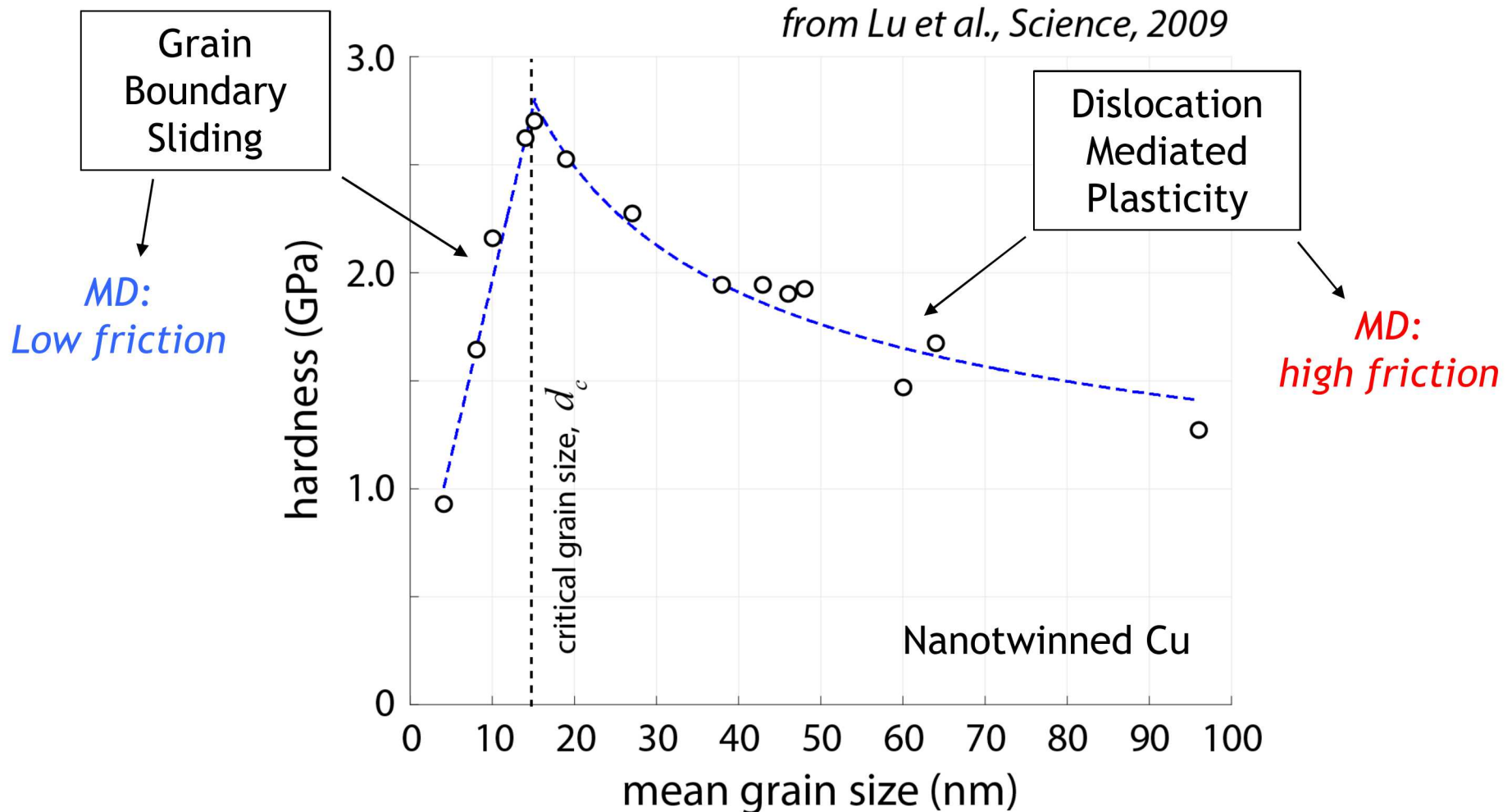
Contact diameters $\sim 10\text{s } \mu\text{m}$
Shear layers $\sim 10\text{s nm}$ thick



For metals, the friction coefficient can be described as the ratio of interface shear strength and bulk hardness:

$$\mu = \frac{\tau}{H}$$

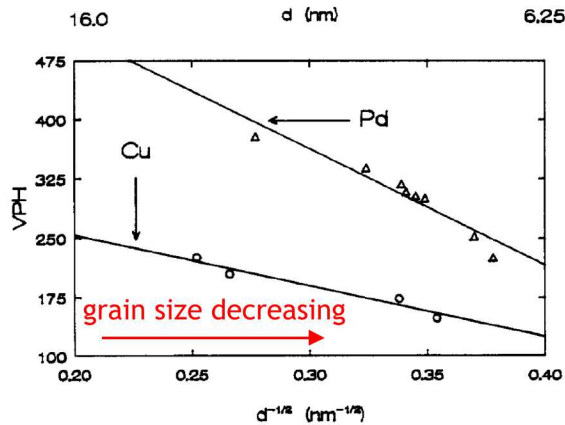
How does grain size affect materials properties?



Hall-Petch breakdown occurs at about 10 nm (critical grain size)

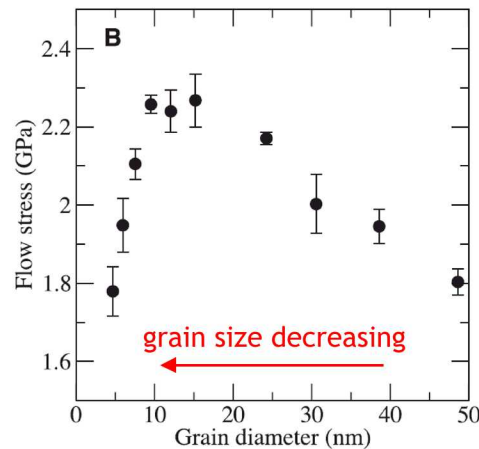
FCC Cu and Pd experiments

Chokshi, et al., Scripta Mat., 1989



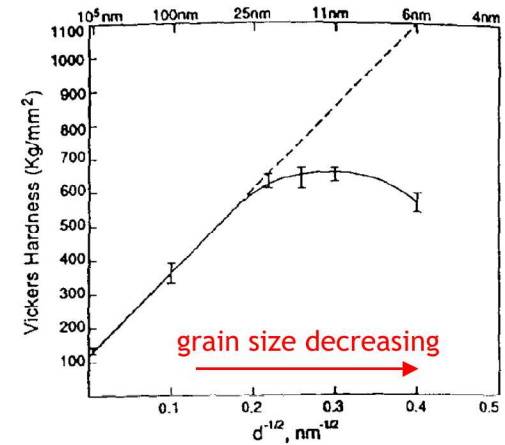
FCC Cu simulations

Schiötz and Jacobsen, Science, 2003



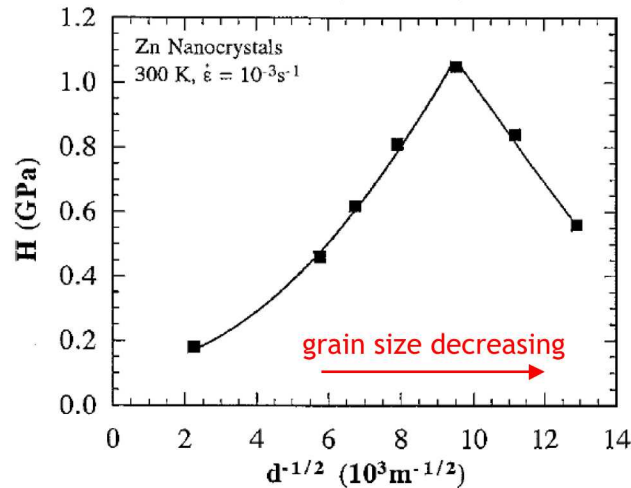
FCC Ni experiments

Erb, Nanostructured Matls., 1995



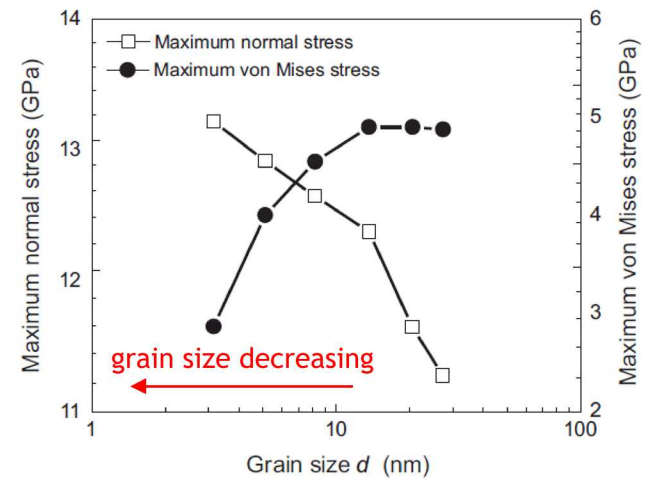
HCP Zn experiments

Conrad and Narayan, App. Phys. Lett., 2002

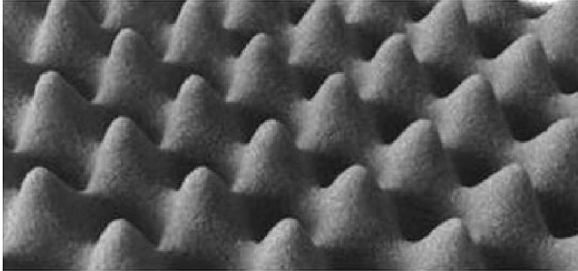


BCC Ta simulations

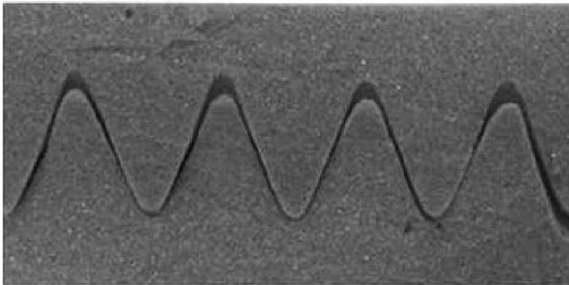
Tang, et al., Mat. Sci. & Eng. A, 2013



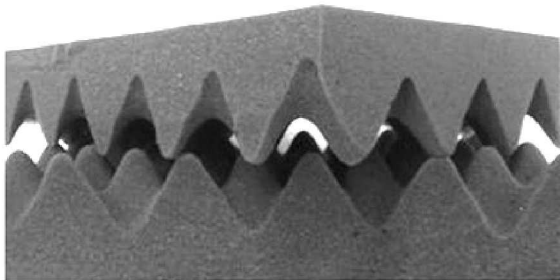
Atomic lattices have periodicity



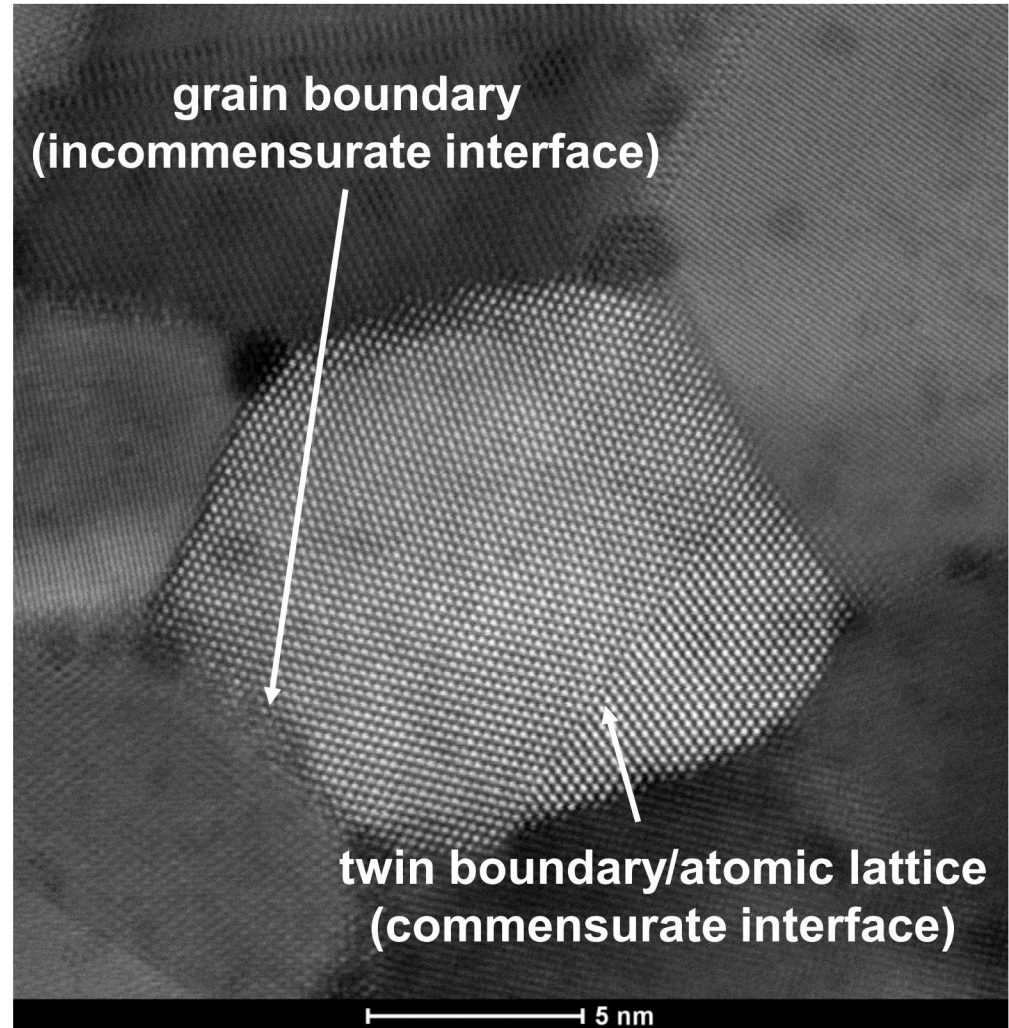
Commensurate interface
(higher energy barrier/higher μ)



Incommensurate interface
(lower energy barrier/low μ)

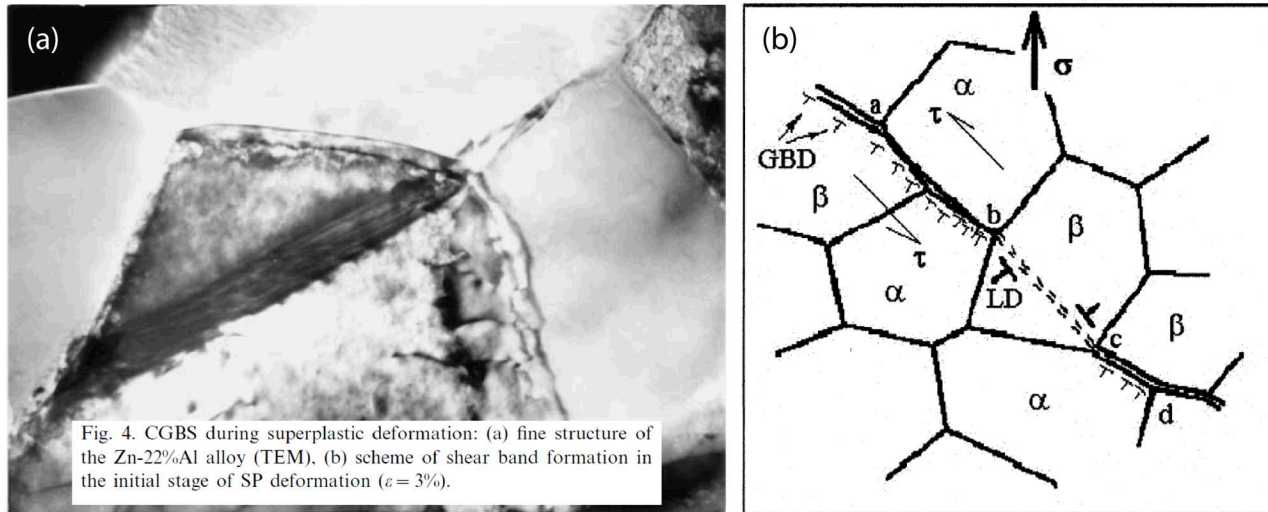


High resolution TEM

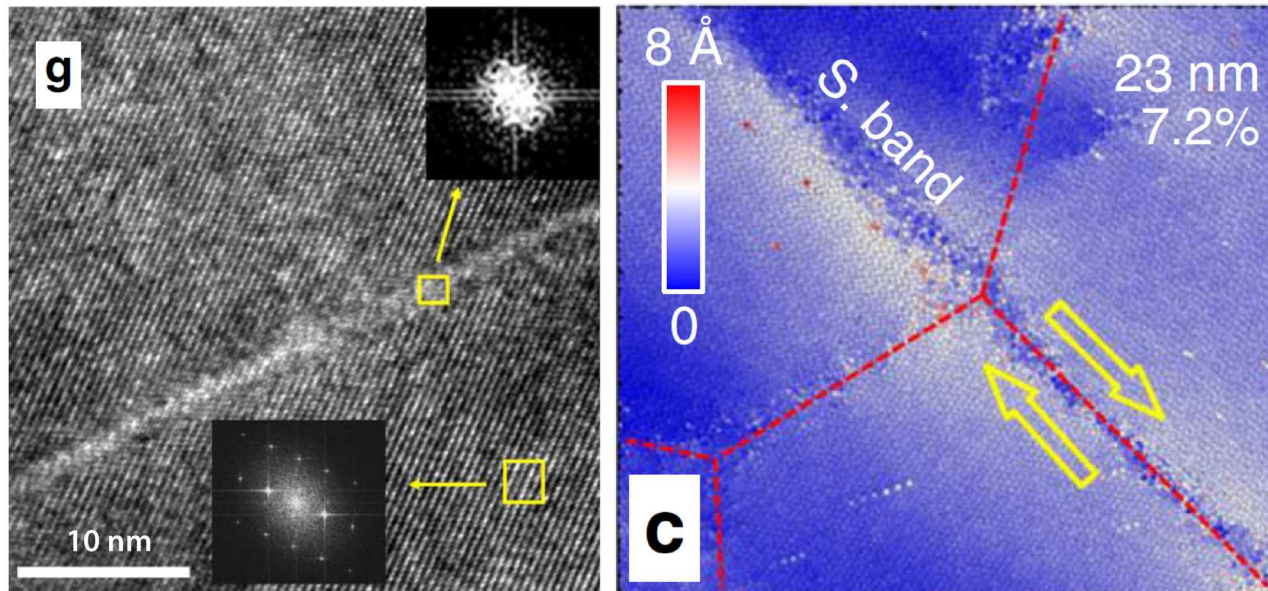


Evidence of Intragranular Amorphization

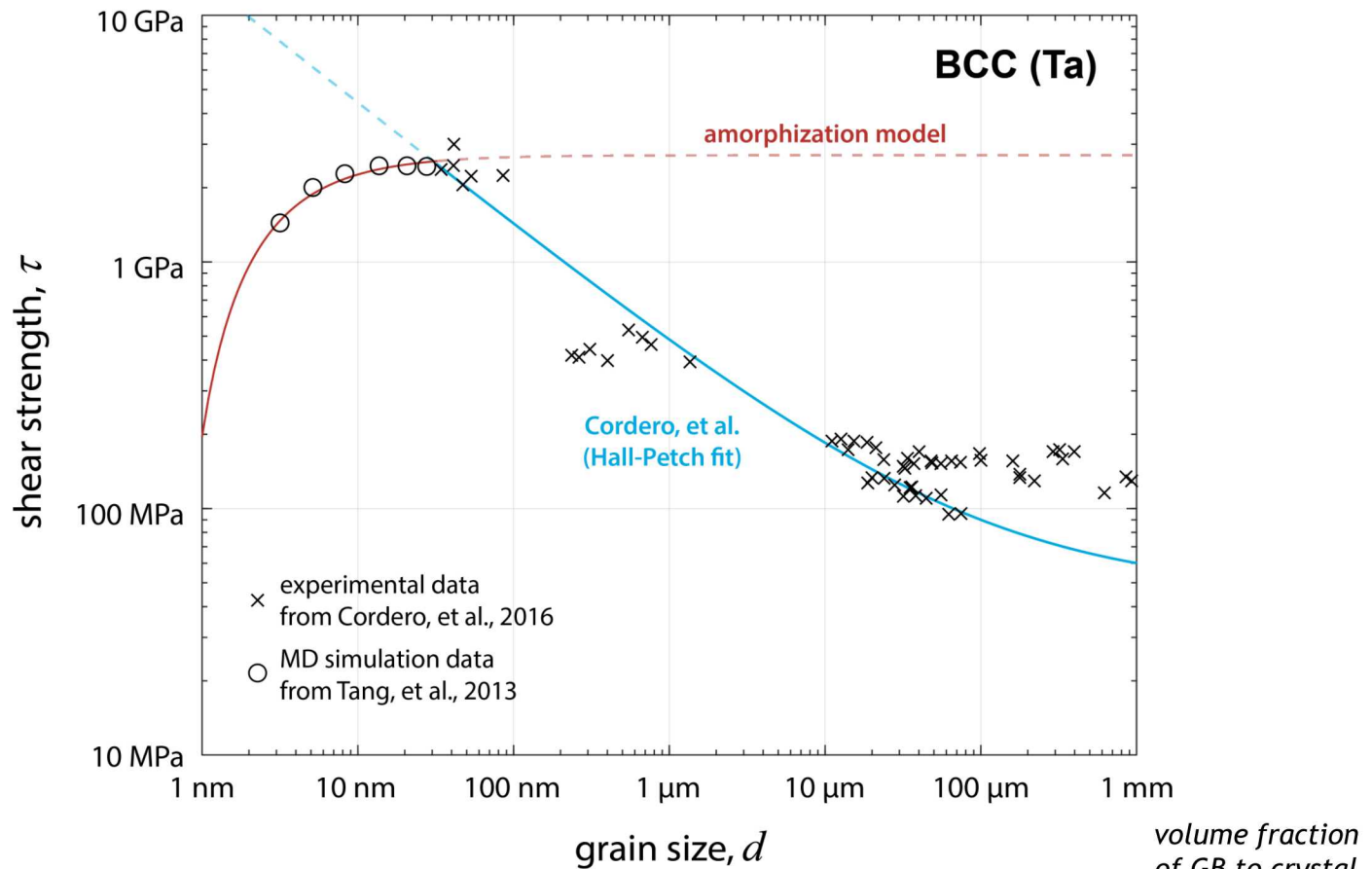
From Kaibyshev, *Mat. Sci. & Eng.*, 2002:



From Luo, *et al.*, *Nat. Comm.*, 2019:



Grain Size Dependent Strength of Metals



Amorphization model:
Chandross and Argibay (in peer review)

$$\tau(d) = L \left(1 - \frac{T}{T_m} \right) \frac{\rho_L}{M} \left(\frac{d - \delta}{d} \right)^3 - \frac{kT}{b^3}$$

heat of
fusion

homologous
temperature

amorphous metallic
atomic volume

thermal
energy

volume fraction
of GB to crystal

Rosenhain and Ewen predict amorphous

THE INTERCRYSTALLINE

BY WALTER ROSENHAIN,

DONALD EWEN,
(BOTH OF THE NAT

IN their first paper on the subject they have put forward what they consider to be in favour of an hypothesis of crystalline cohesion in metals which may be briefly termed the "amorphous" hypothesis, that the crystals of which a metal is composed are "cemented" together by an amorphous or non-crystalline material. The substance of the metal or the intercrystalline layer is regarded as being at least closely analogous to an undercooled liquid which does not meet one another in various

Slip at Grain Boundaries and Grain Growth in Metals

By N. F. MOTT,
H. H. Wills Physical Laboratory, Bristol

(b) The observed fact that at the melting point the slip is the same as that which would be given by a monomolecular layer of liquid appears in the theory as an *accident*. The mechanism of flow in liquid aluminium cannot be anything like that sketched here, because the temperature dependence is 10–20 times smaller; the viscosity of liquid metals depends on temperature according to the formula

$$\sigma = \sigma_0 e^{W/kT},$$

where W is of the order of the latent heat of fusion (Frenkel 1946).

We must therefore modify our hypothesis. Let us suppose that the elementary act which allows slip to occur is the *disordering* of atoms round each island where it is good. The free energy F necessary to do this will approach zero at the melting point and nL at the absolute zero of temperature; here L is the latent heat of fusion per atom. At any other temperature, let us assume F to be given by

$$F = nL(1 - T/T_M),$$

where T_M is the temperature of melting. Let us assume also, since the disordering will result in a slip through a distance a , that a stress σ will decrease or increase F by $\pm \frac{1}{2} \sigma n \omega a$. Then the rate of slip is now

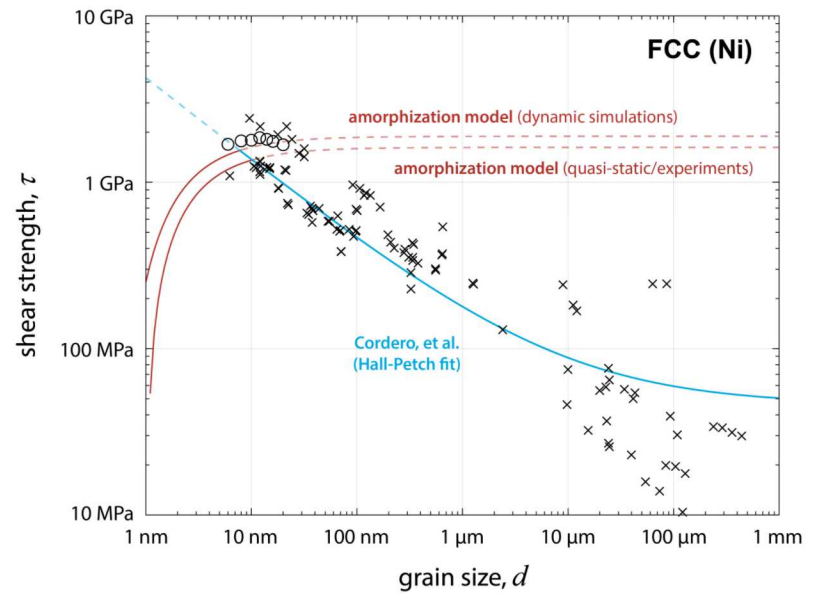
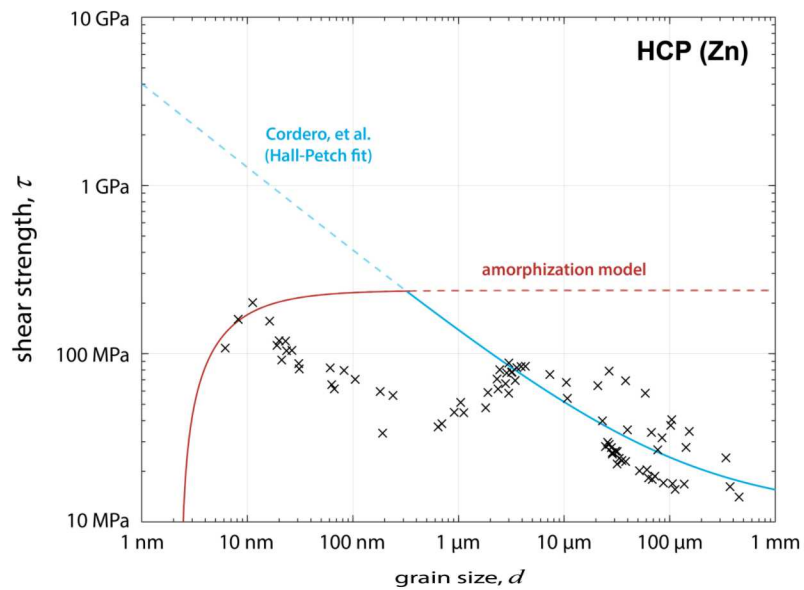
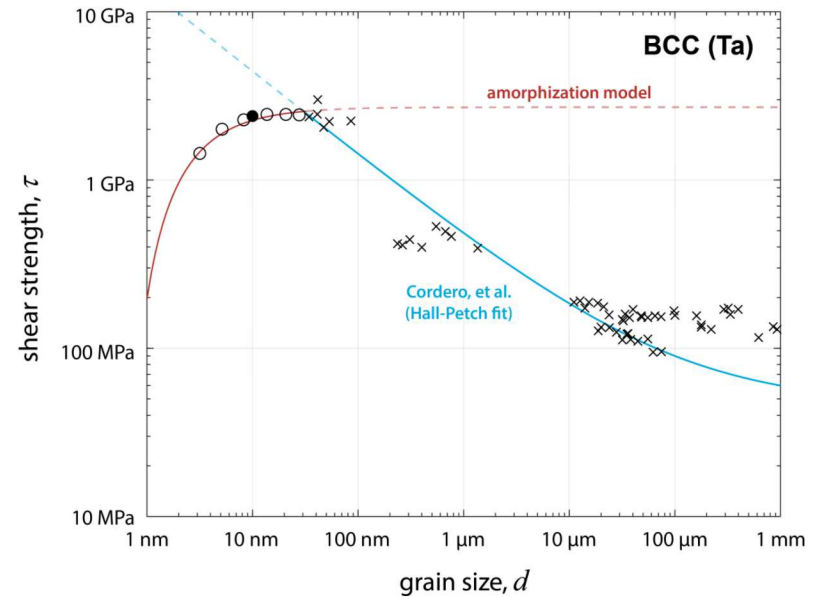
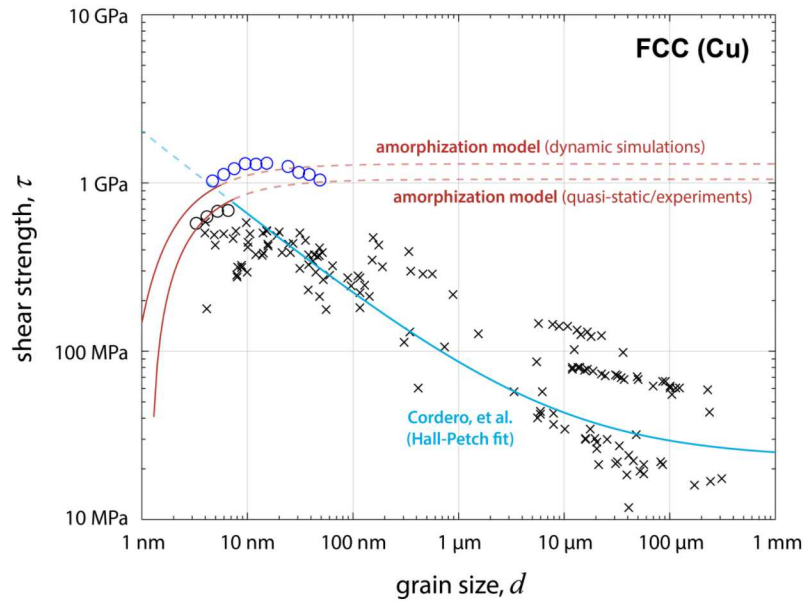
$$v = 2va \exp \left\{ -nL(1 - T/T_M)/kT \right\} \sinh (\sigma n \omega a / 2kT),$$

which for small σ reduces to

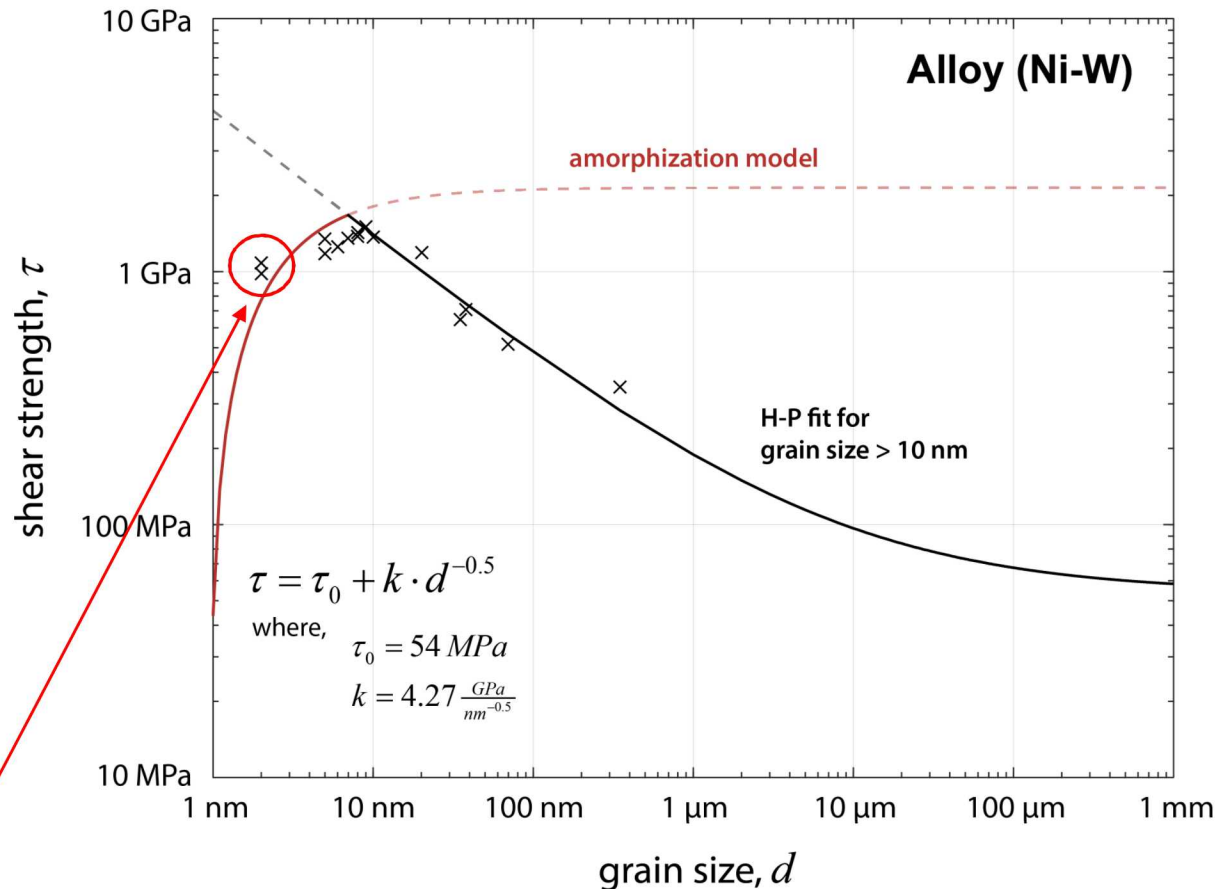
$$v = \frac{va^2 n \omega \sigma}{kT} \exp \left(\frac{nL}{kT_M} \right) \exp \left(\frac{-nL}{kT} \right).$$

Mott provided a theory of slip based on the formation of disordered, liquid-like islands of atoms

The model works well for many other metals



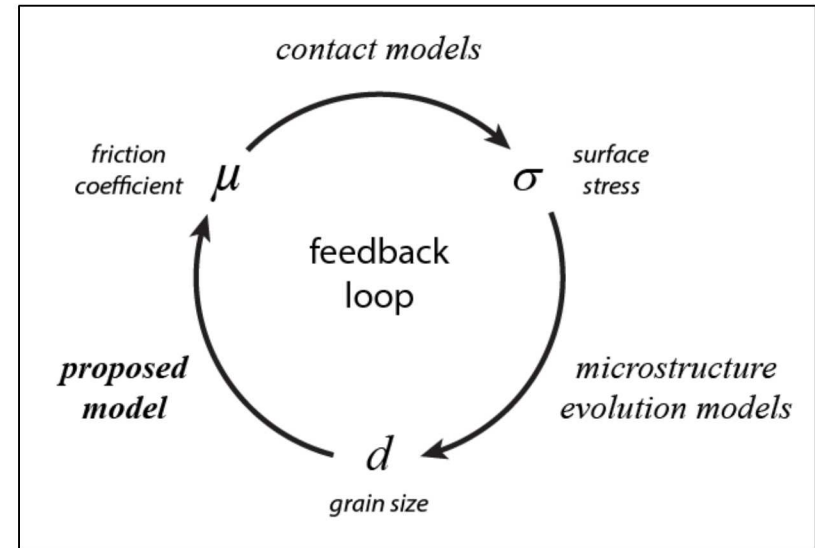
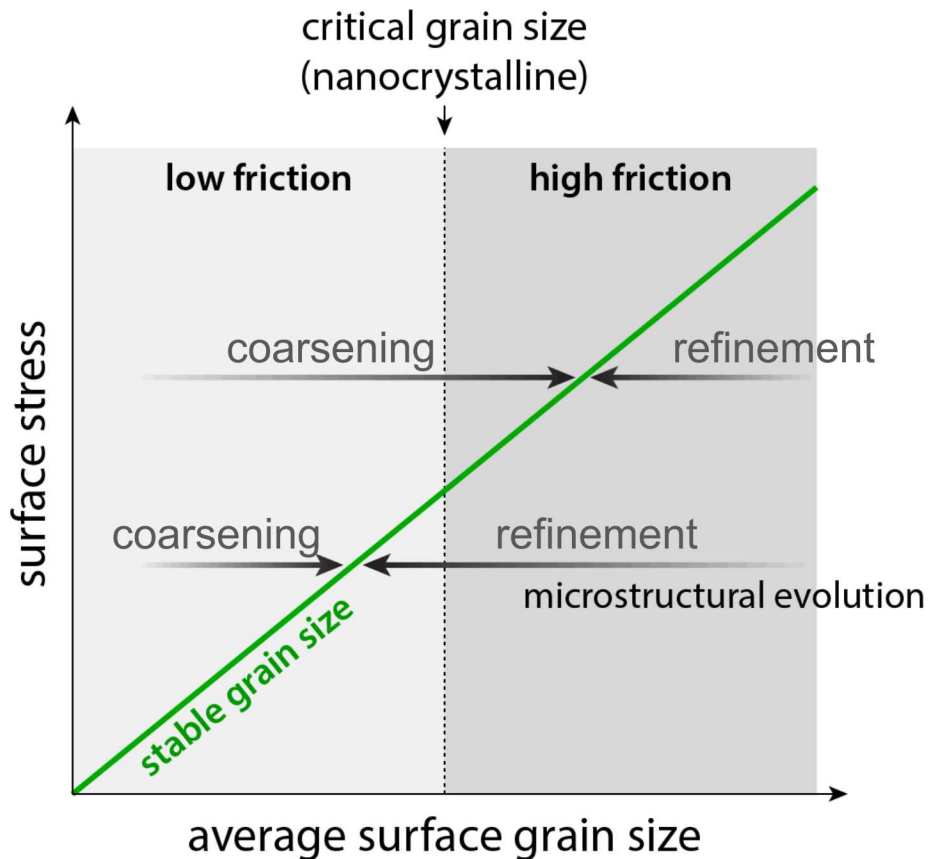
Amorphization model also works for alloys



Keep in mind:

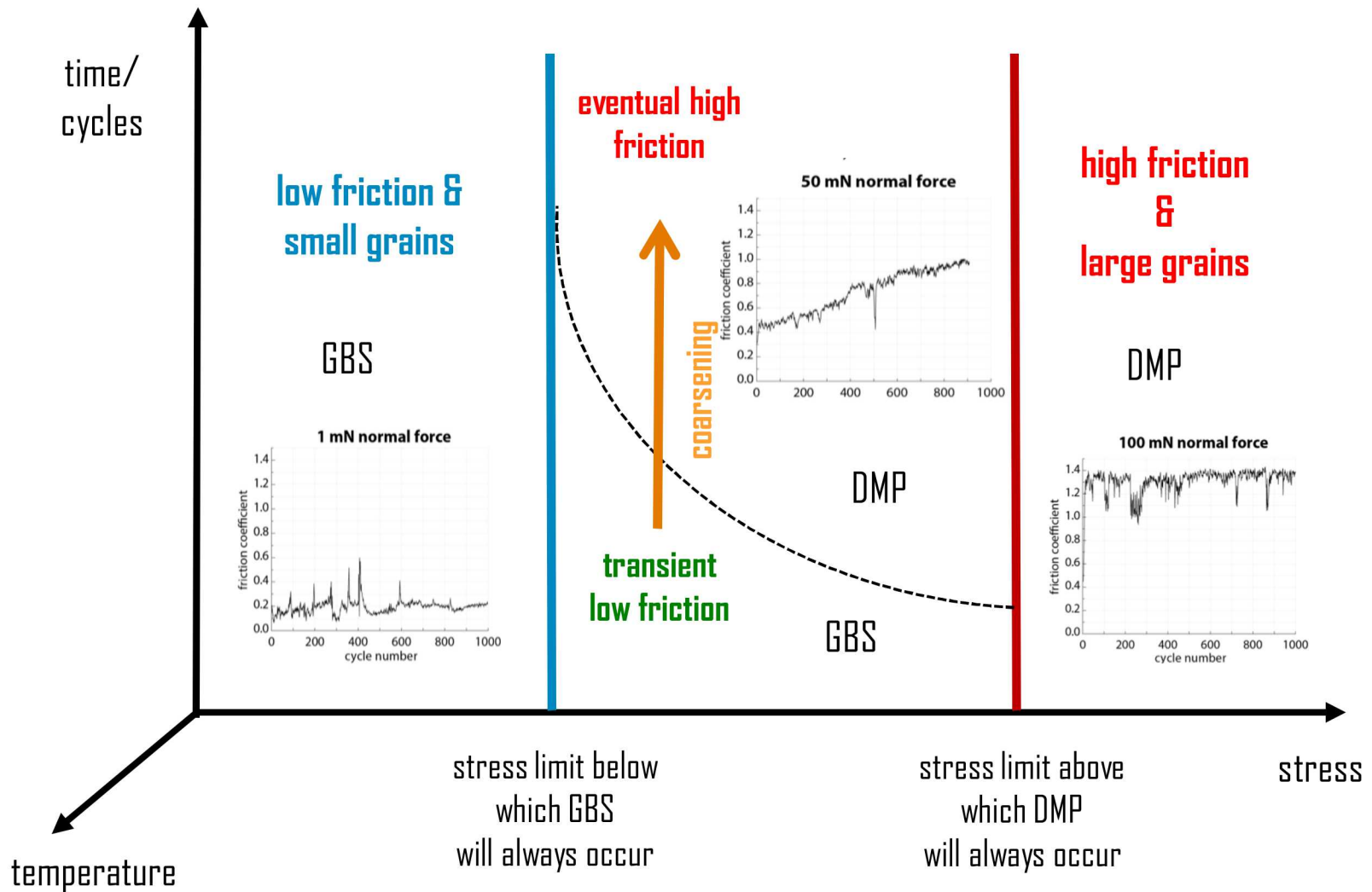
1. Grain size estimates are particularly difficult and exhibit higher uncertainty in the 1-10 nm range, and
2. these errors are relatively small compared to typical H-P fit for far coarser grain size (e.g. earlier pure metal data)

Proposed stress-dependent steady-state (asymptotic) grain size

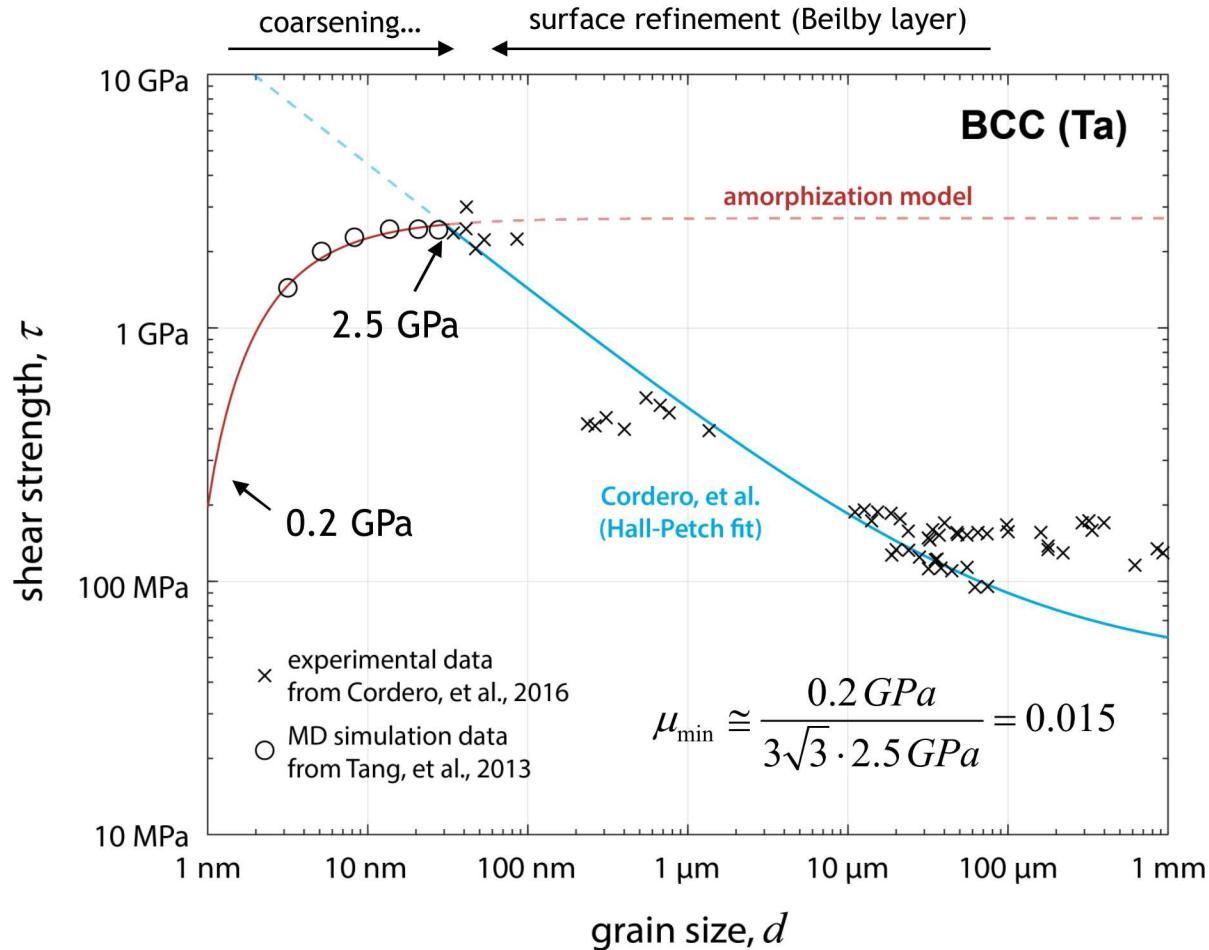


- Effective refinement from recrystallization (Zener & Holloman, 1944; McQueen et al., 1967)
- Known in rocks and ice cores (Derby et al., 1992)
- Recently extended to metals under severe plastic deformation (Pougis et al., 2014)

Generalized friction regimes map for metals

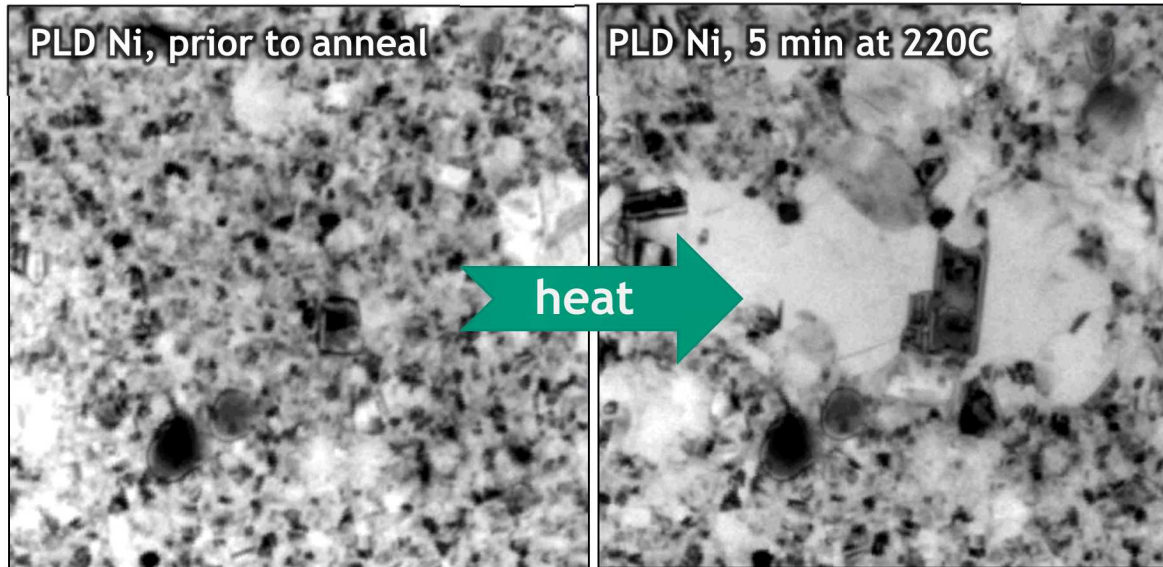


What is the lowest friction condition?



Bowden & Tabor: $\mu = \frac{\tau}{H} \xrightarrow{\text{Tabor}} \mu = \frac{\tau}{3 \cdot \sigma_Y} \xrightarrow{\text{Von Mises}} \mu = \frac{\tau(d_{\text{surface}})}{3\sqrt{3} \cdot \tau(d_{\text{bulk}})}$

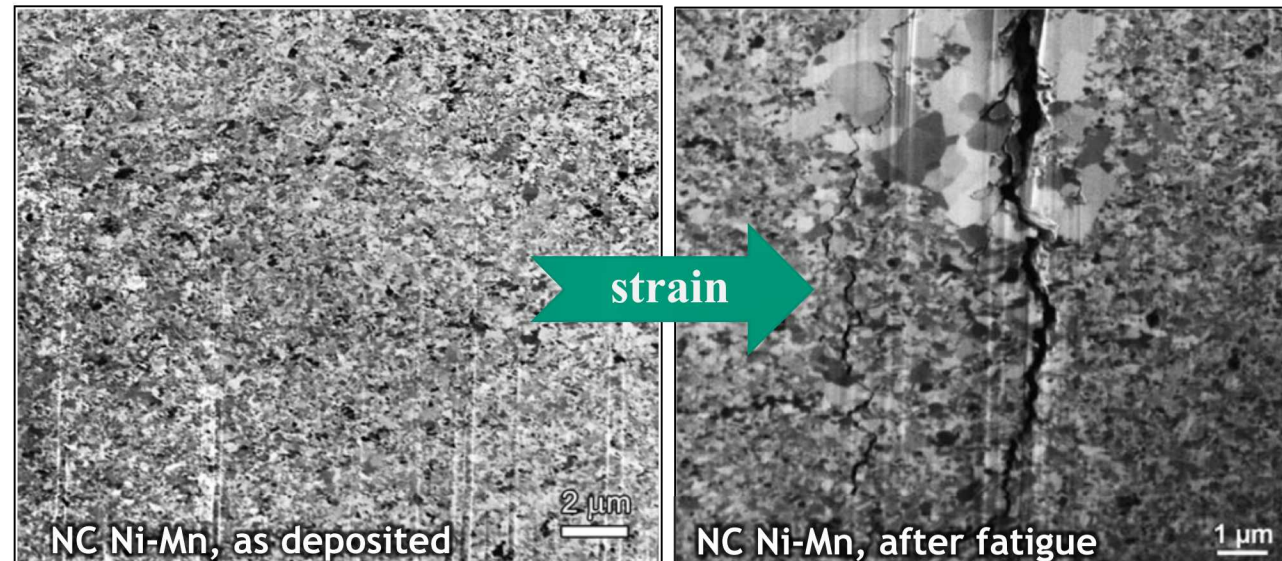
The challenge of grain size instability...



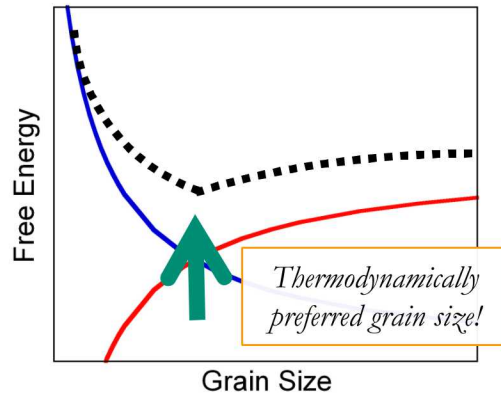
Grain growth in nanocrystalline metals is driven

Thermally...

...and Mechanically



Slide courtesy of C. Schuh and H. Murdoch (MIT)



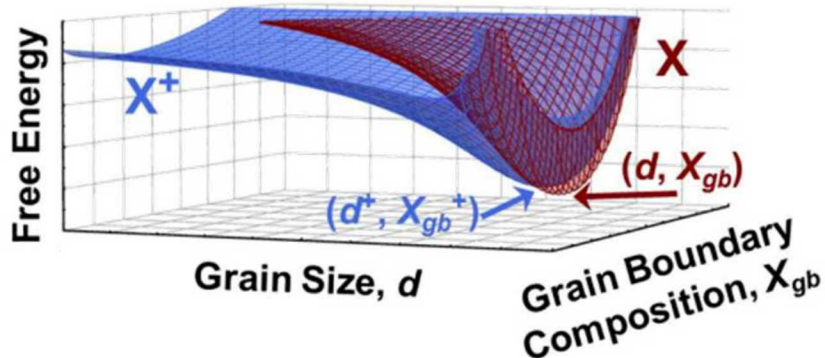
RNS Model:

Binary metal alloys exist possessing highly (intrinsically?) thermodynamically stable nanocrystallinity.

Solute segregation at grain boundaries:

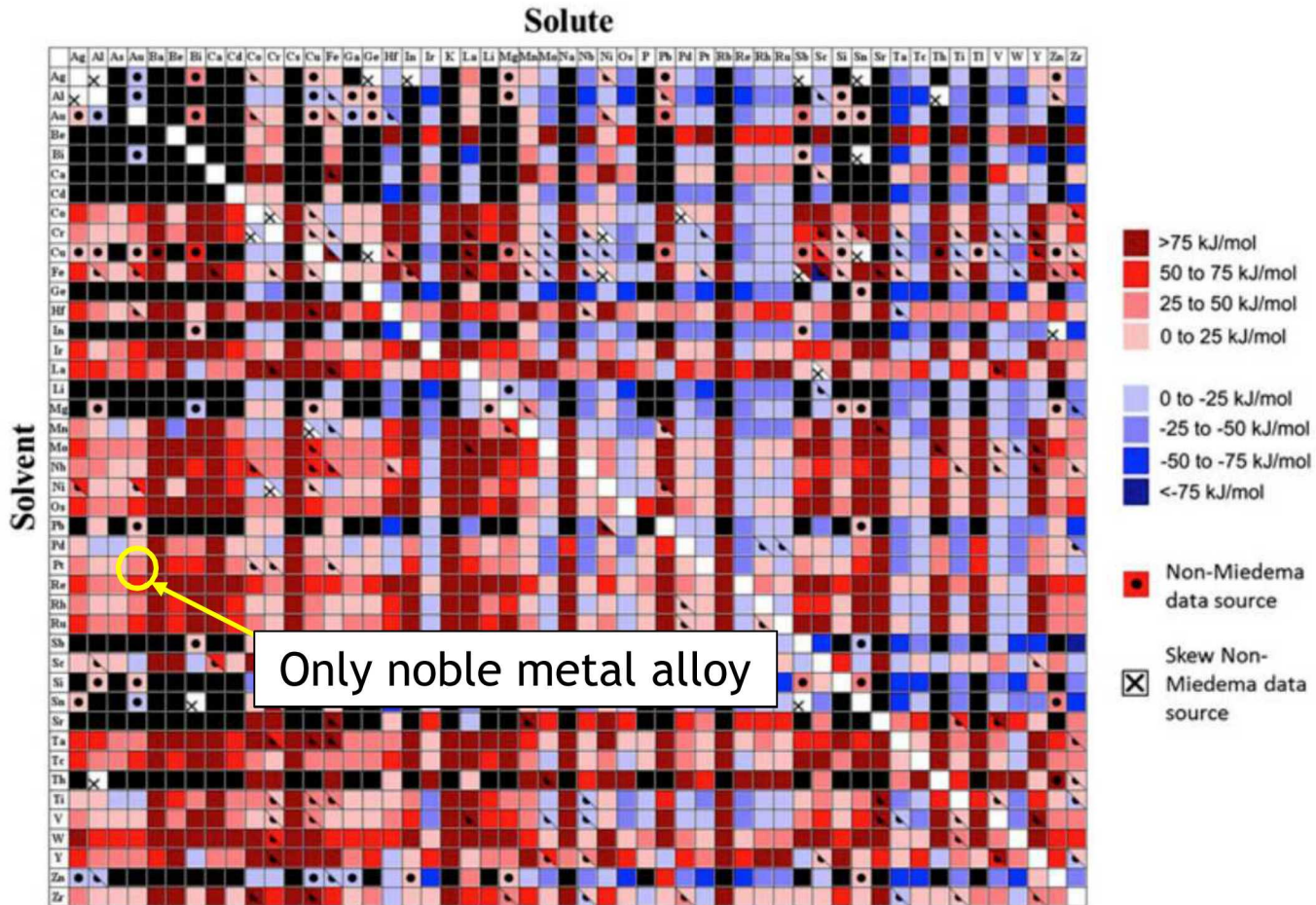


$$dG = \left[\gamma - \frac{N_{\beta}}{A} \Delta G_{seg} \right] dA$$



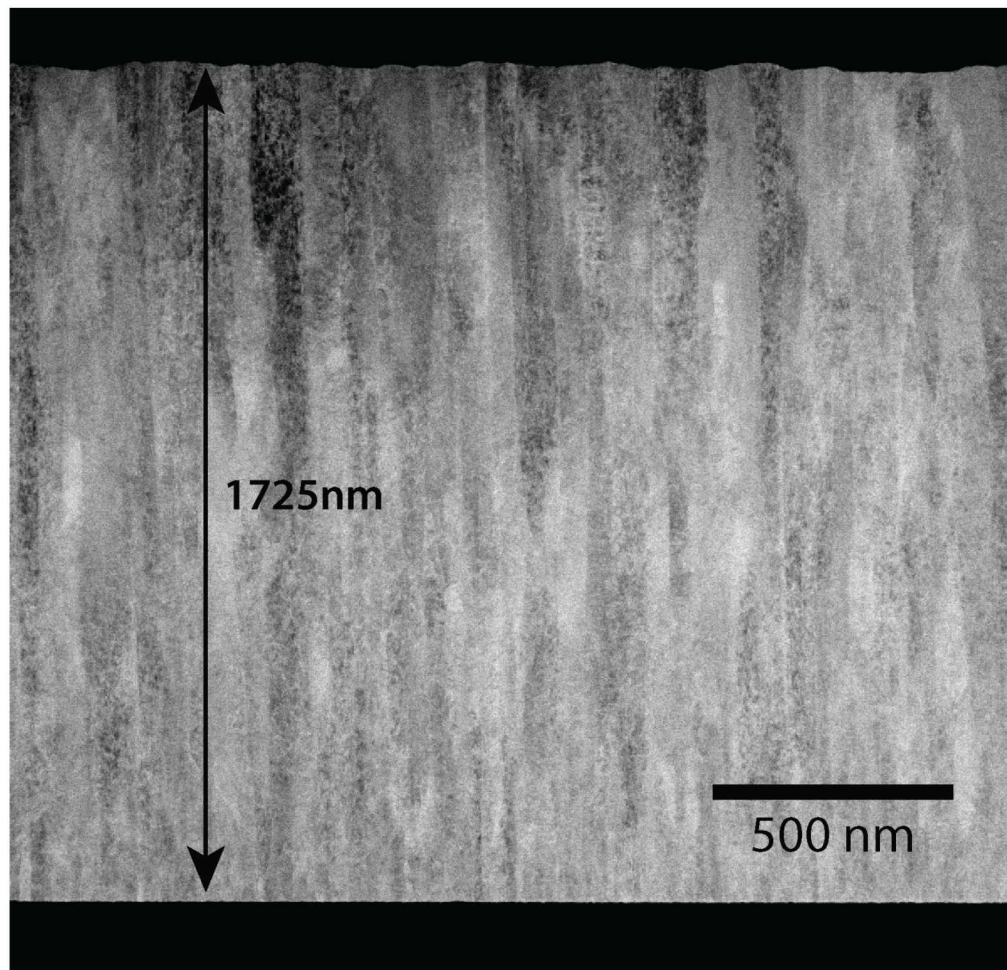
References:

- T. Chookajorn, et al., *Science*, 2012
- Kirchheim, *Acta Materialia*, 2002
- Weissmuller, J. *Materials Research*, 1994
- D.S. Gianola et al., *Acta Materialia*, 2006

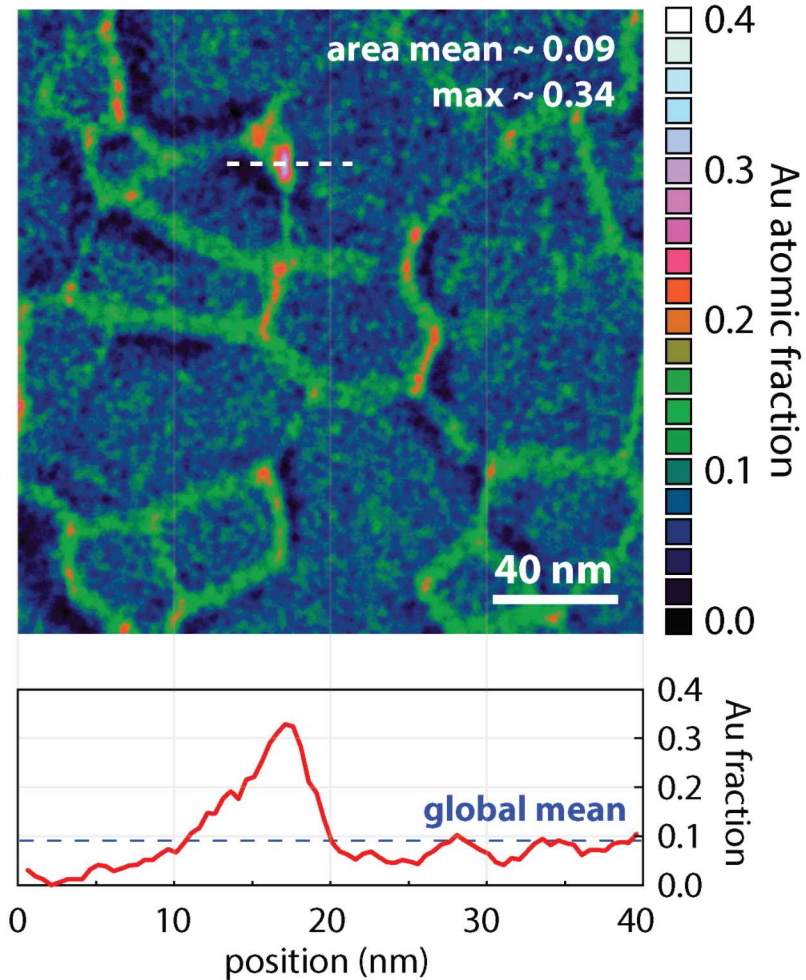


Murdoch & Schuh, *J. Mater. Res.* 2013.

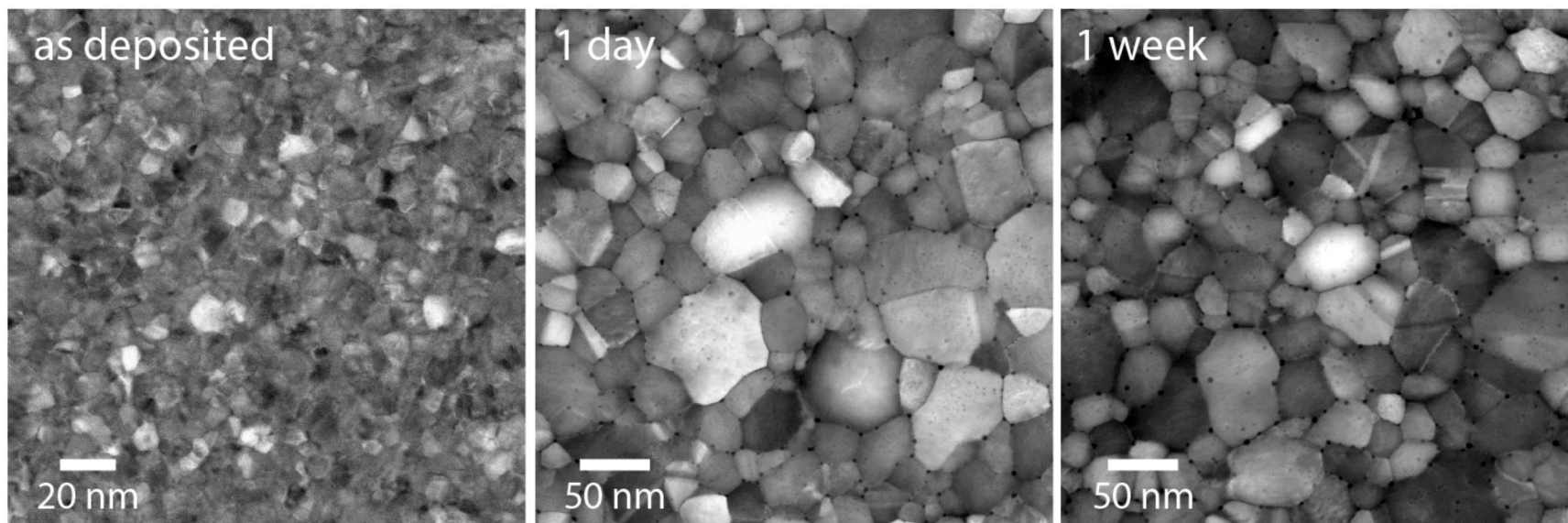
A) PtAu cross-section



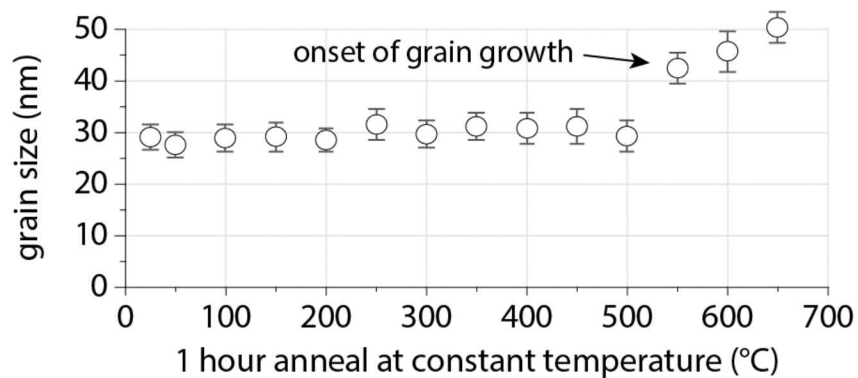
B) Au atomic fraction



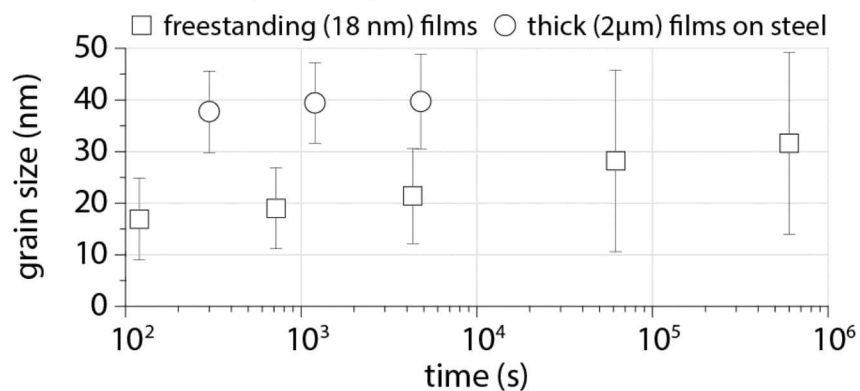
We synthesized coatings of Pt with 10% Au



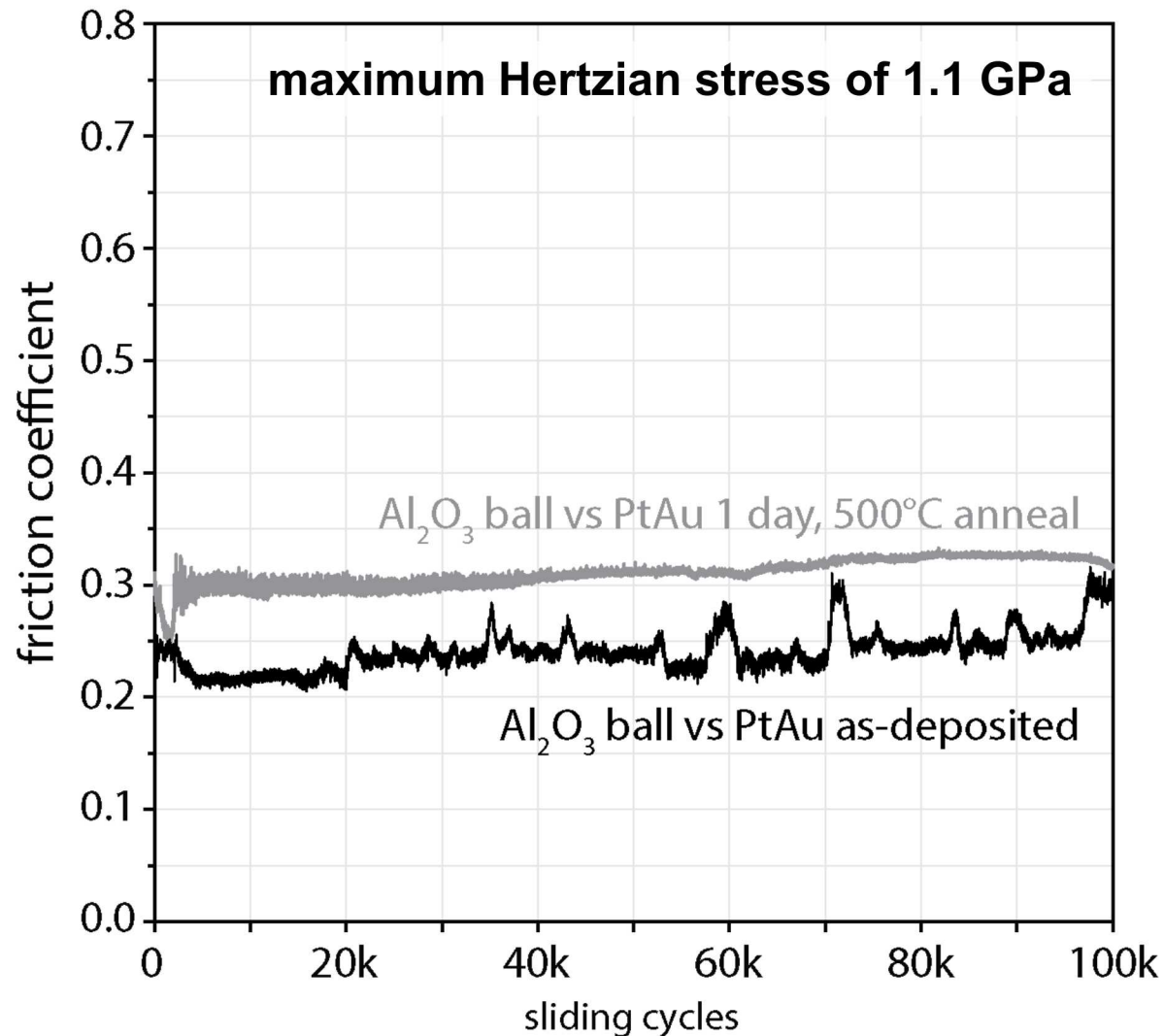
XRD in situ grain growth



TEM in situ grain growth at 500°C

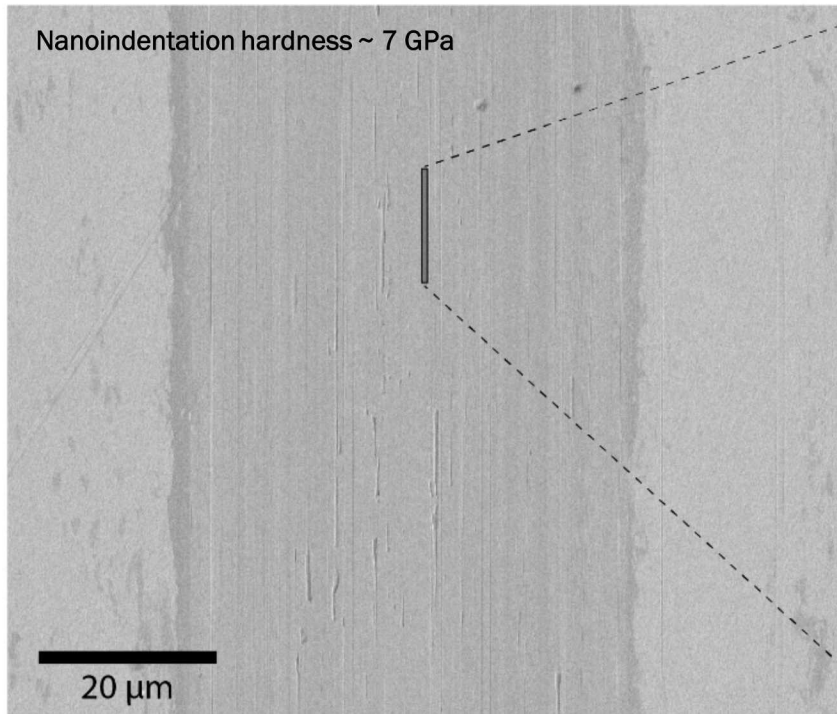


Stable nanocrystalline PtAu exhibited long-lived low friction
(sliding against sapphire)

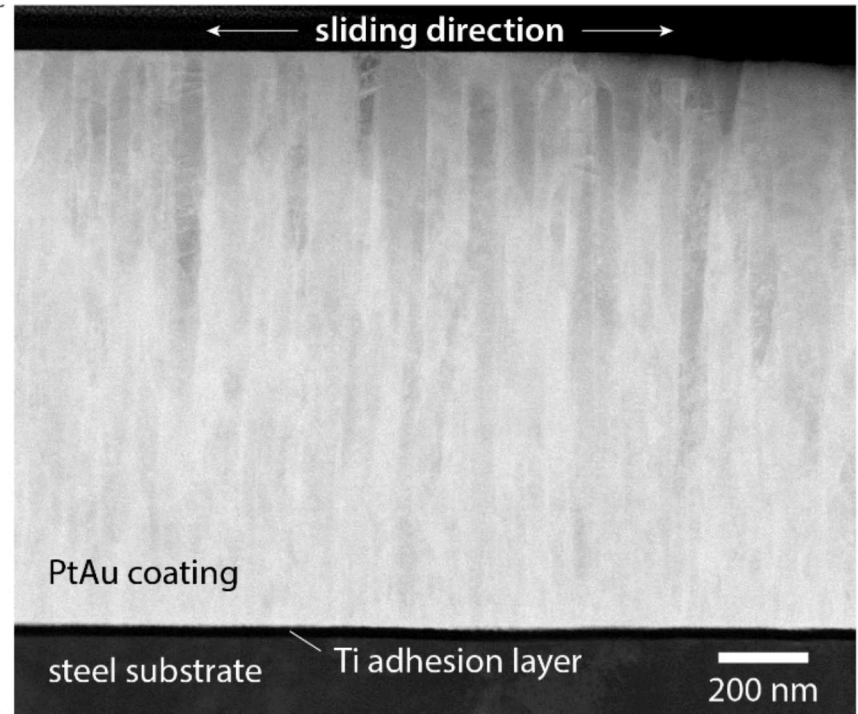


No evidence of microstructural evolution after prolonged tests

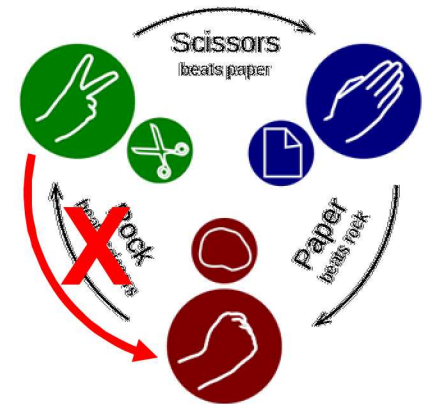
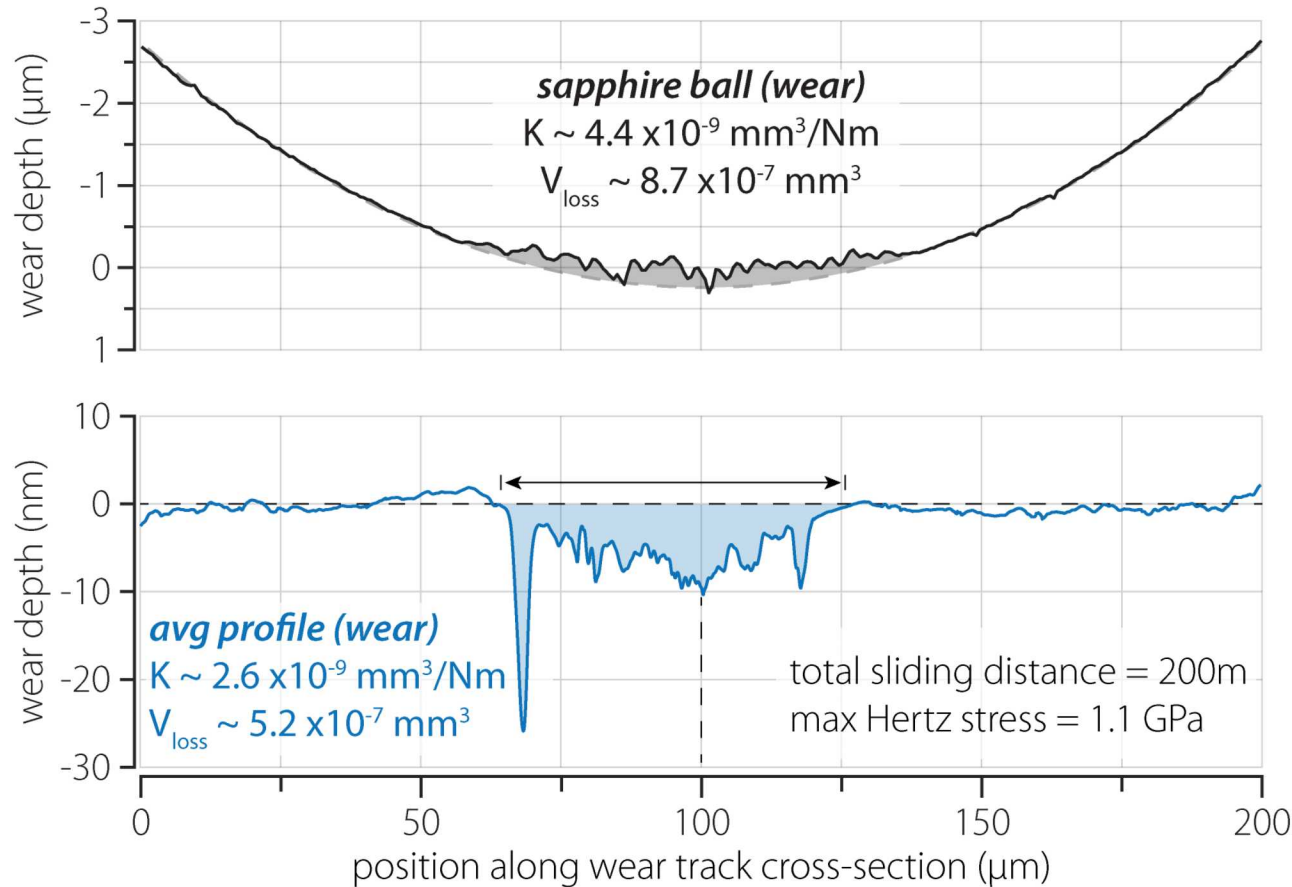
SEM of PtAu wear track after 100k passes



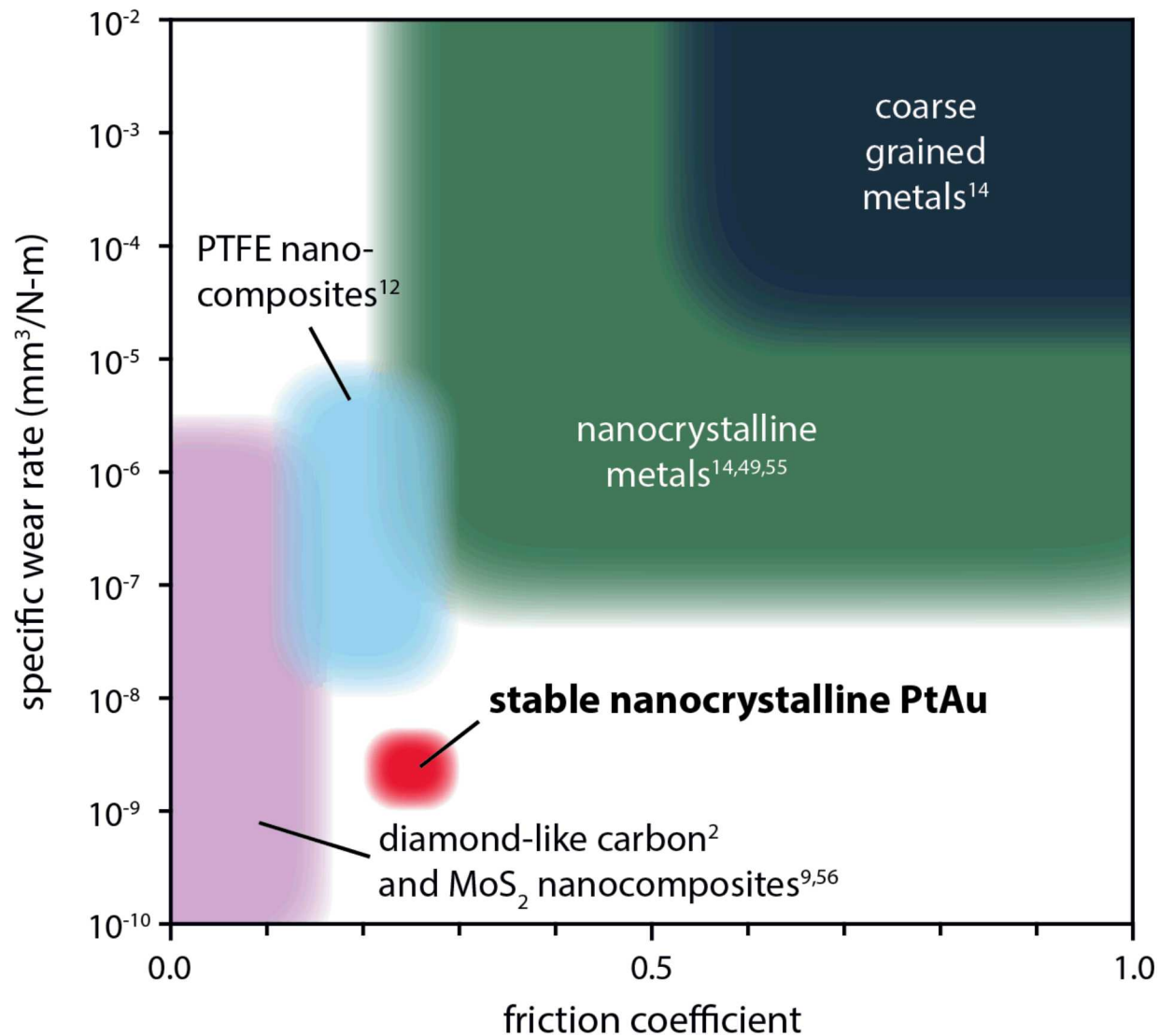
TEM cross-sections of wear track

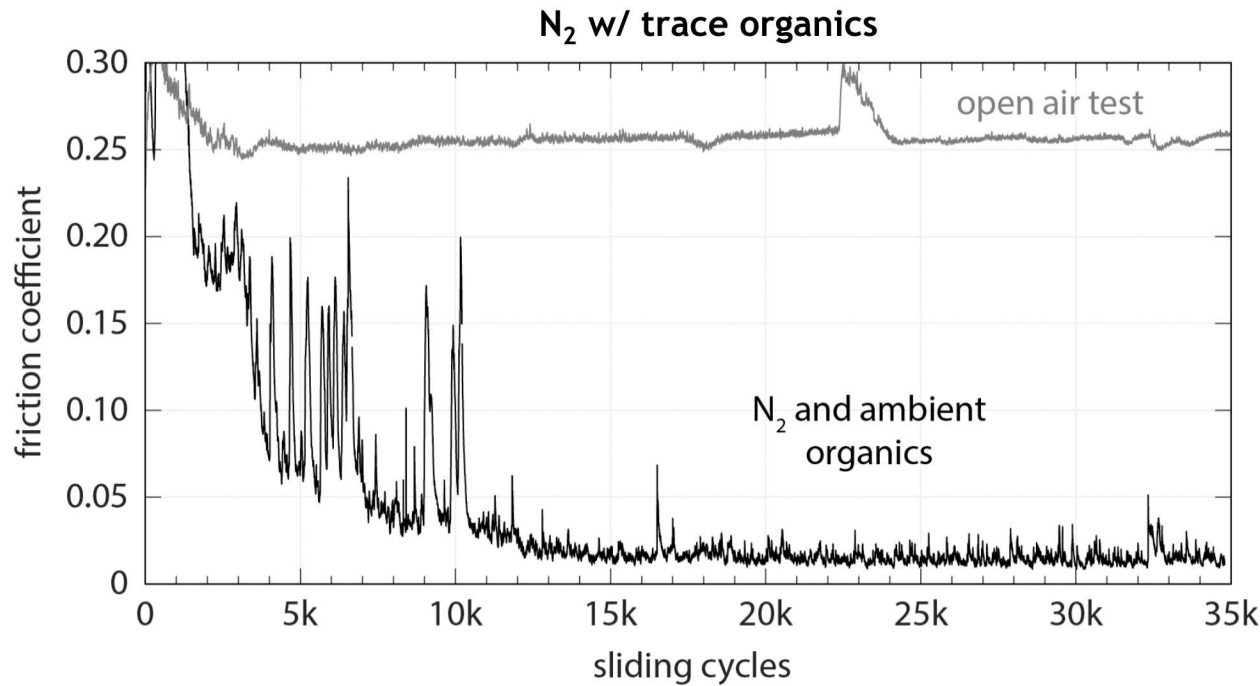


No evidence of microstructural evolution
(maximum Hertzian stress of 1.1 GPa)



... when scissors beat rock!





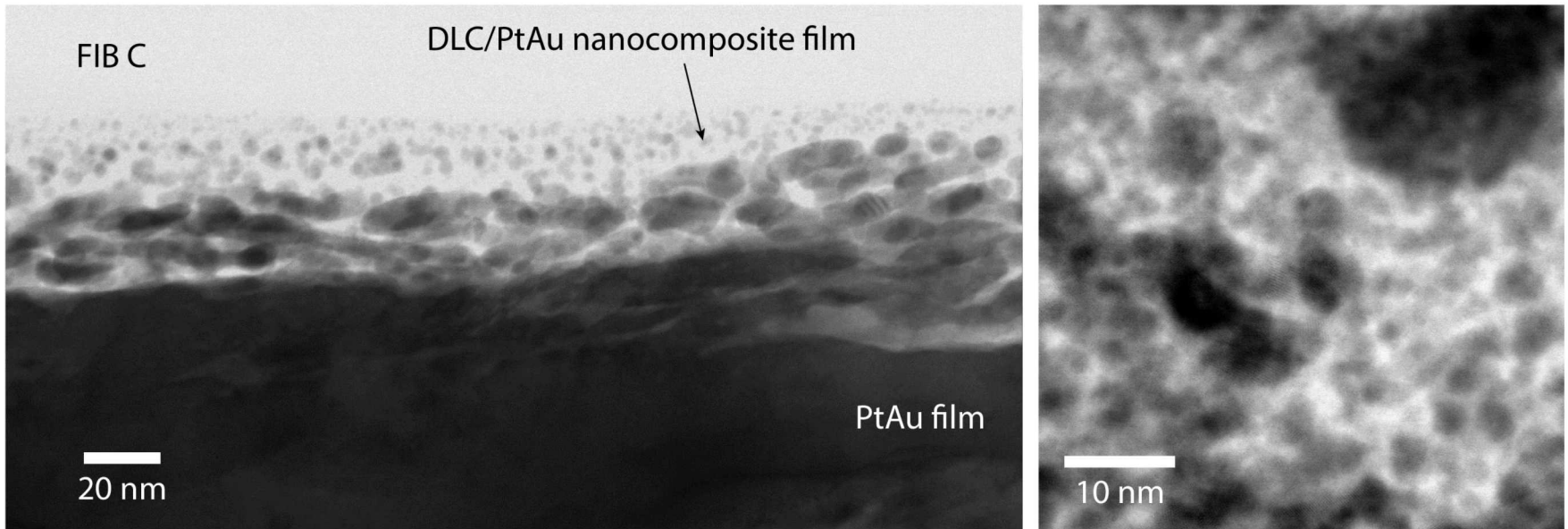
Compounds Identified

trichloroethylene
acetic acid
benzaldehyde
tridecane
phenol
acetophenone
2-heptyl-1,3-dioxolane
butylated hydroxytoluene
benzoylformic acid
phenylmaleic anhydride

(20-100 ng/L)

*... in situ tribo-chemical formation of DLC films from “thin air”!
(Albuquerque is at 6,000ft/ 1,800m of elevation, after all...)*

TEM Reveals a Nanocomposite Structure (DLC/PtAu Nanoparticles)



confirmed **diamond-like carbon (DLC)** using Raman analysis
with 20% hydrogenation using elastic recoil

Acknowledging the Many Contributors to this Story

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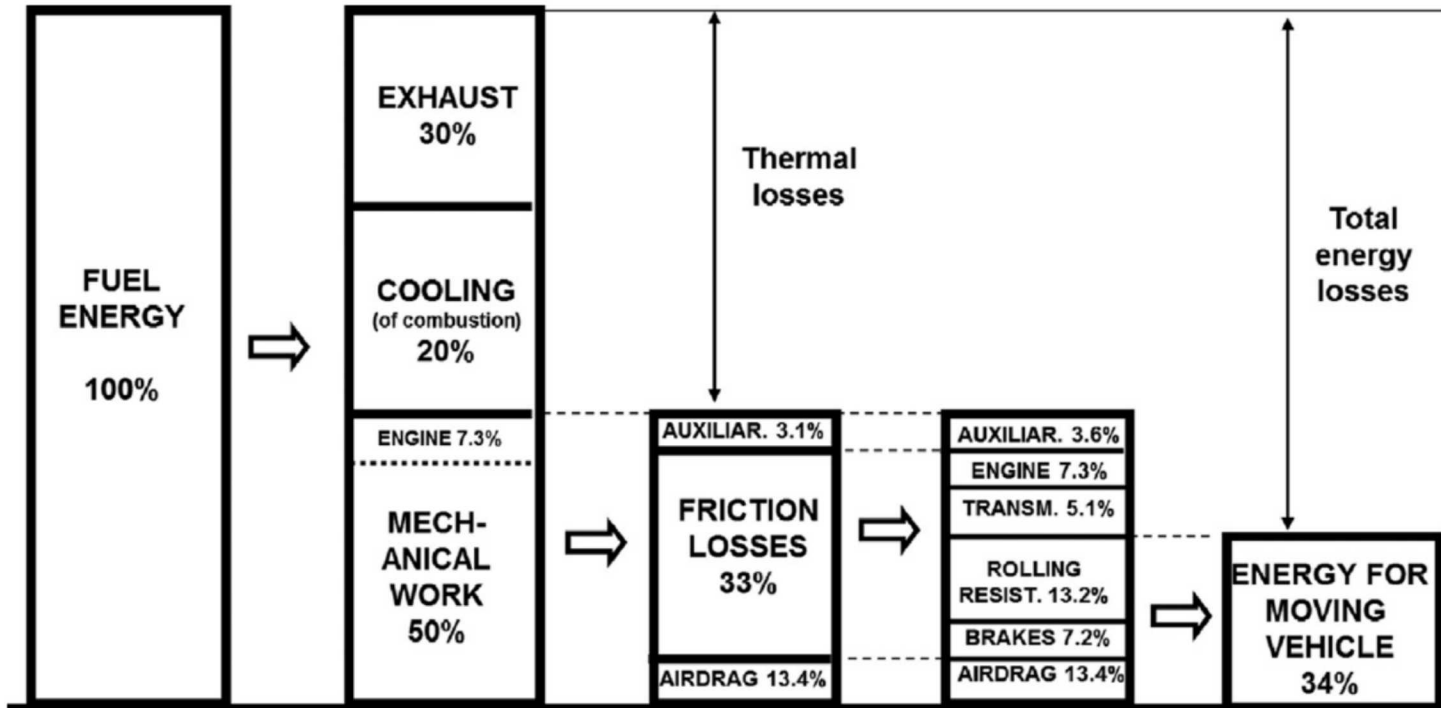
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Special thanks to Dr. Michael Chandross
(who is typically a co-presenter...)

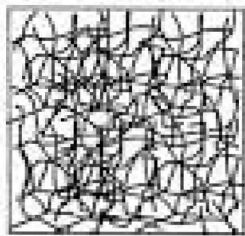
Questions?



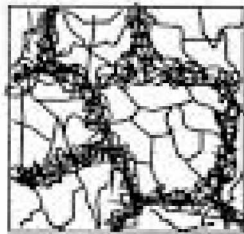
ref: K. Holmberg et al., *Tribo. Int.* 2012

Competition between refinement and coarsening

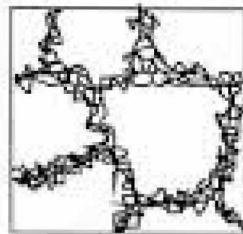
http://engineering.dartmouth.edu/defmech/chapter_2.htm



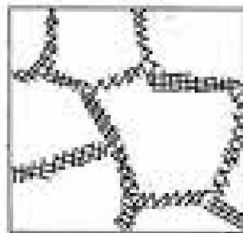
(a) Dislocation tangles



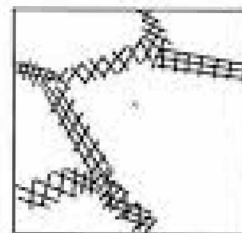
(b) Cell formation



(c) Annihilation of dislocations within cells

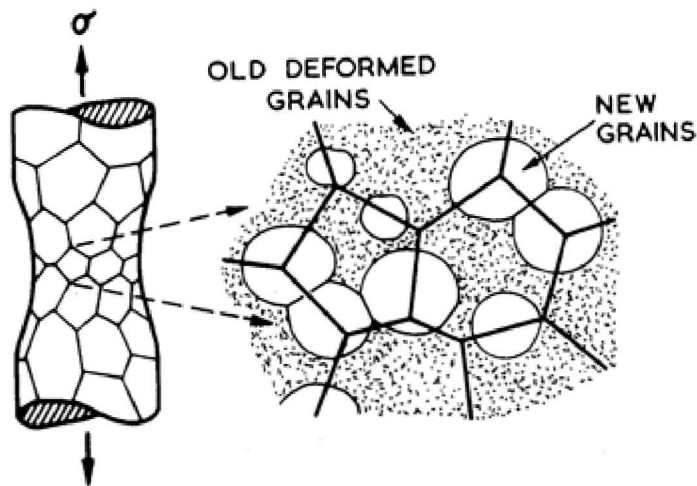


(d) Subgrain formation

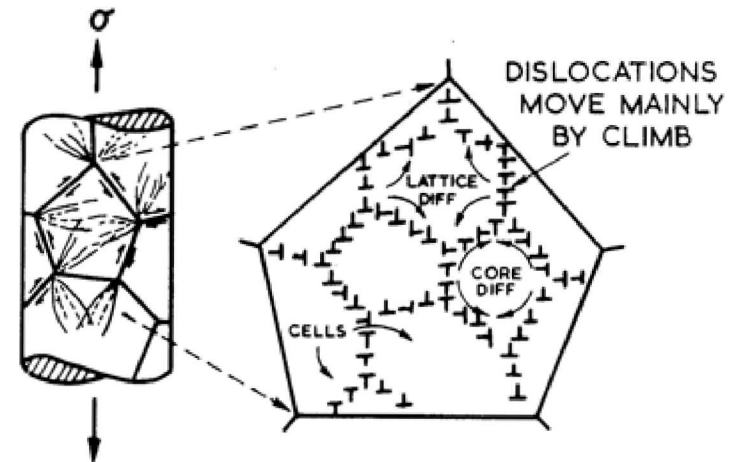


(e) Subgrain growth

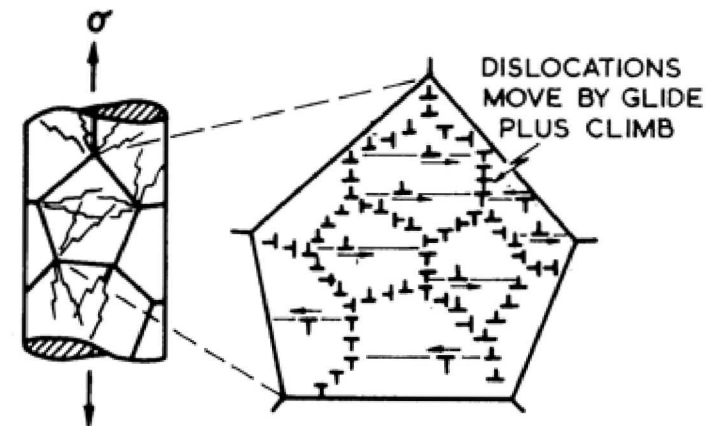
DRX:



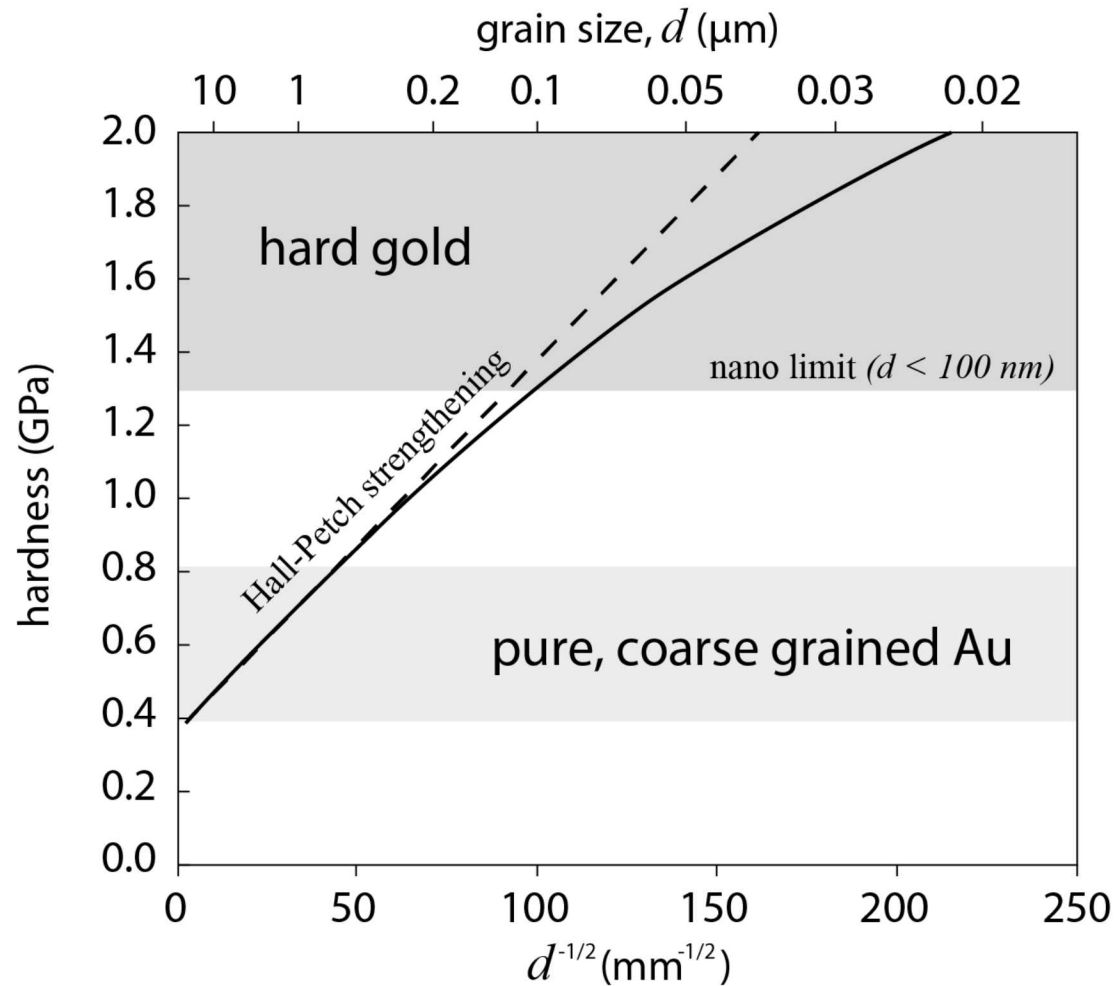
creep:



glide:



Grain-size dependent strength: Hall-Petch

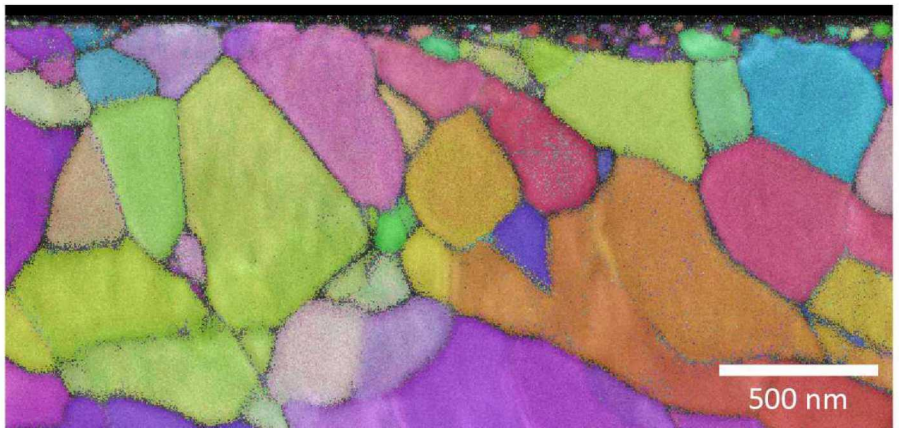
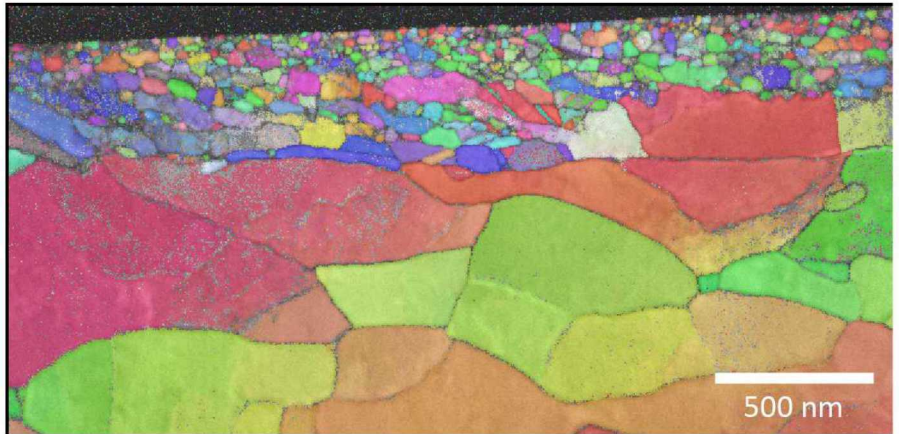
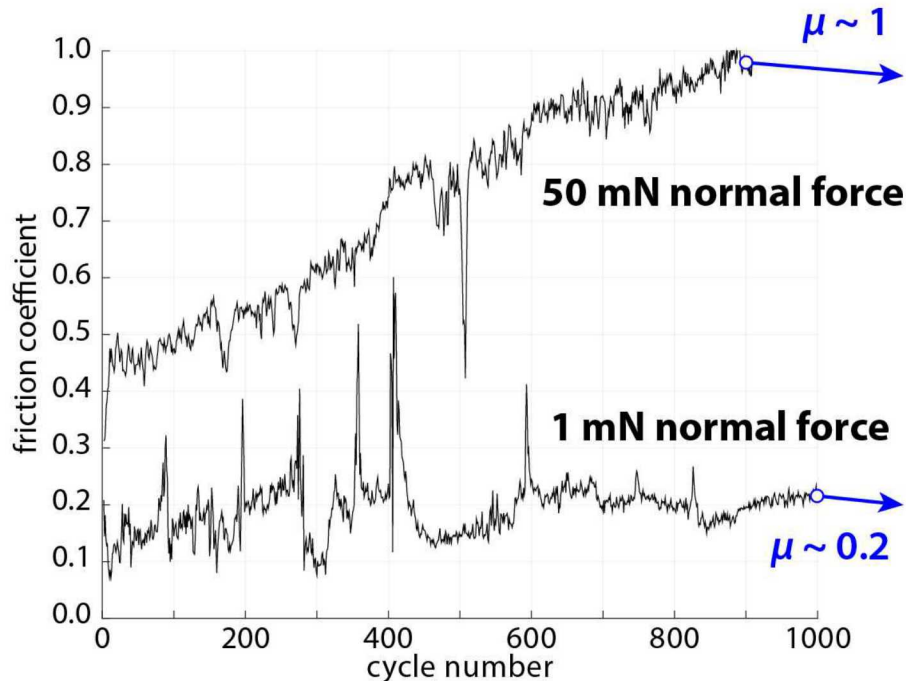


Reference: C. Lo, J. Augis, and M. Pinnel, JAP (1979)

Electron diffraction of high and low friction wear tracks from Au-Au sliding contacts

Electron microscopy of focused ion beam prepared wear track cross-sections

Transmission Kikuchi Diffraction (TKD):
(transmission diffraction performed in an SEM)

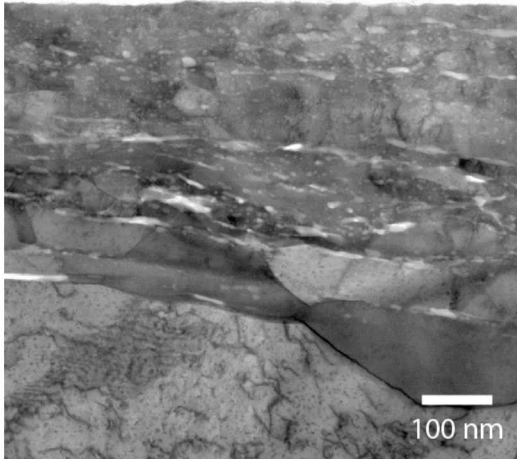


Again we see fine grain size in both cases...
but the low friction case seems smaller.

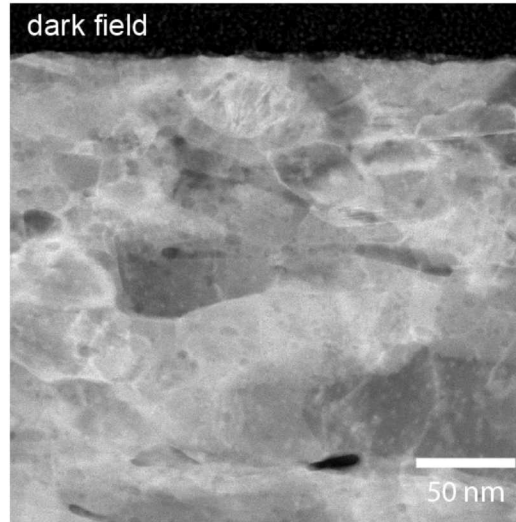
Microstructurally small differences in grain size = BIG difference in friction response

50 mN normal force, $\mu_{\text{final}} \sim 1.0$

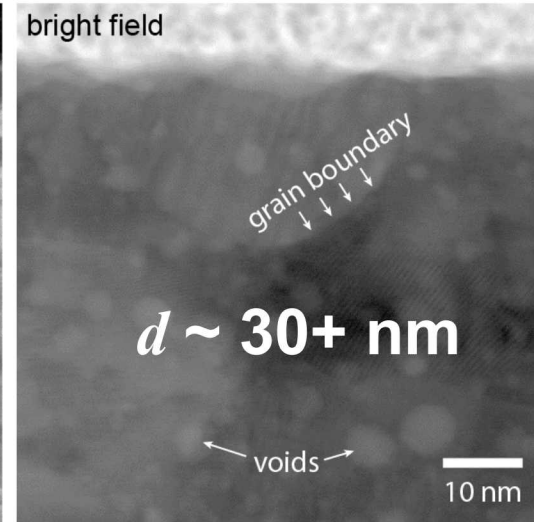
bright field



dark field

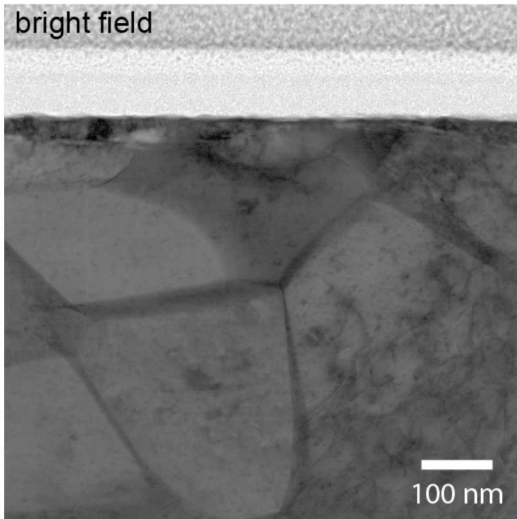


bright field

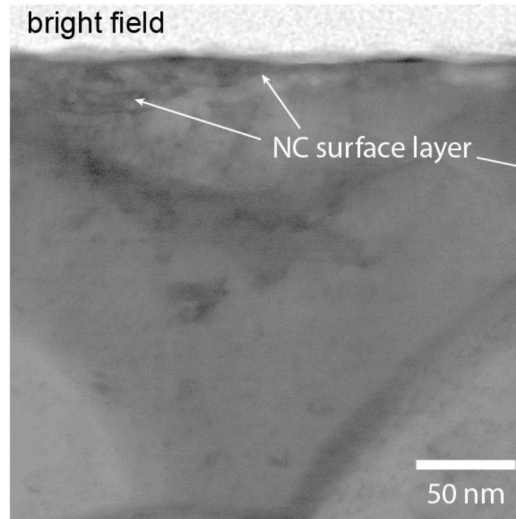


1 mN normal force, $\mu_{\text{ss}} \sim 0.2$

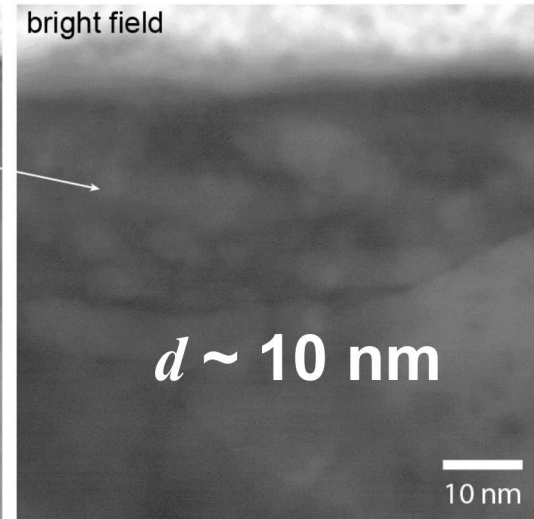
bright field



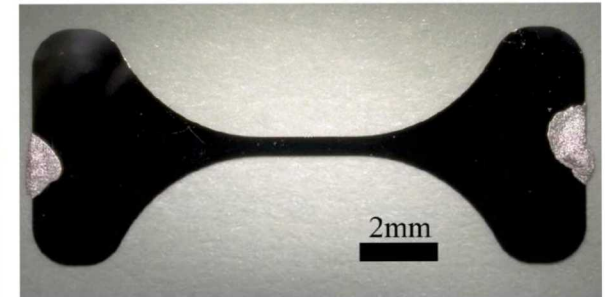
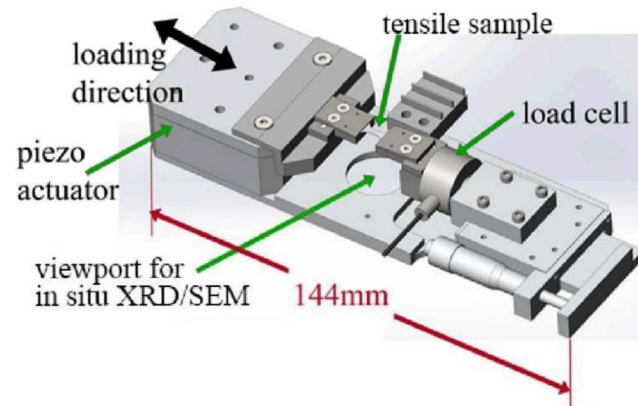
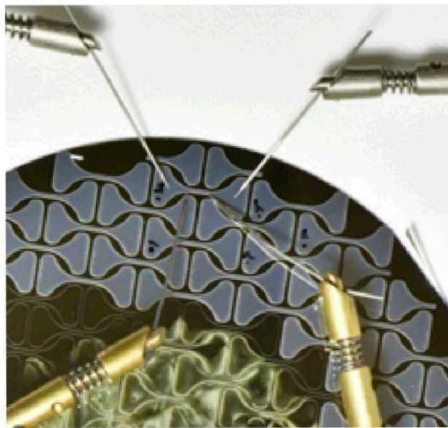
bright field



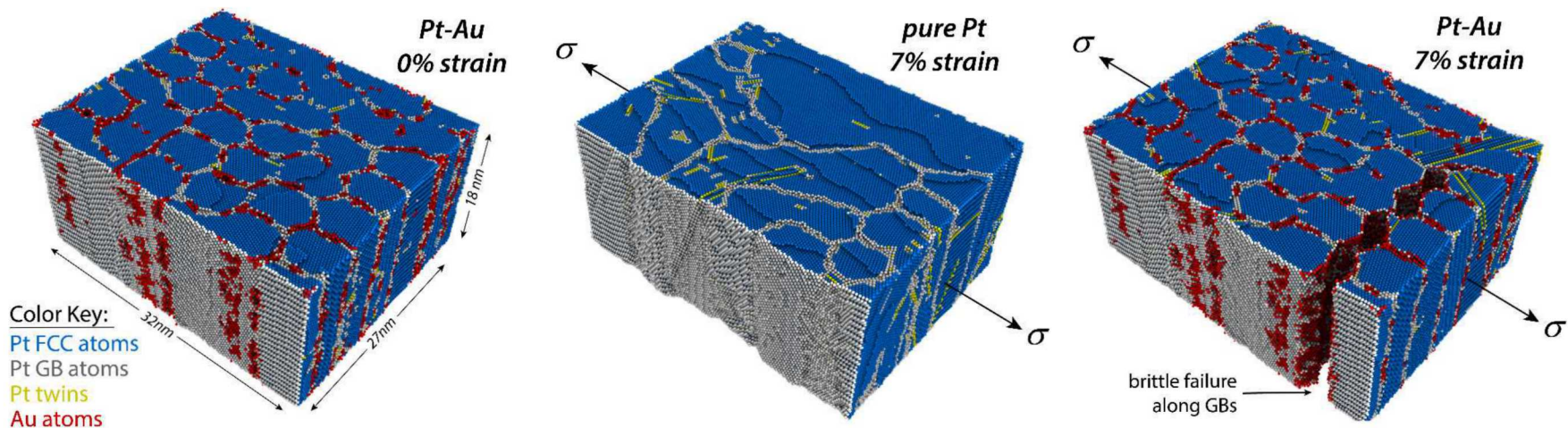
bright field



Tensile Testing (Fatigue and Yield Strength):

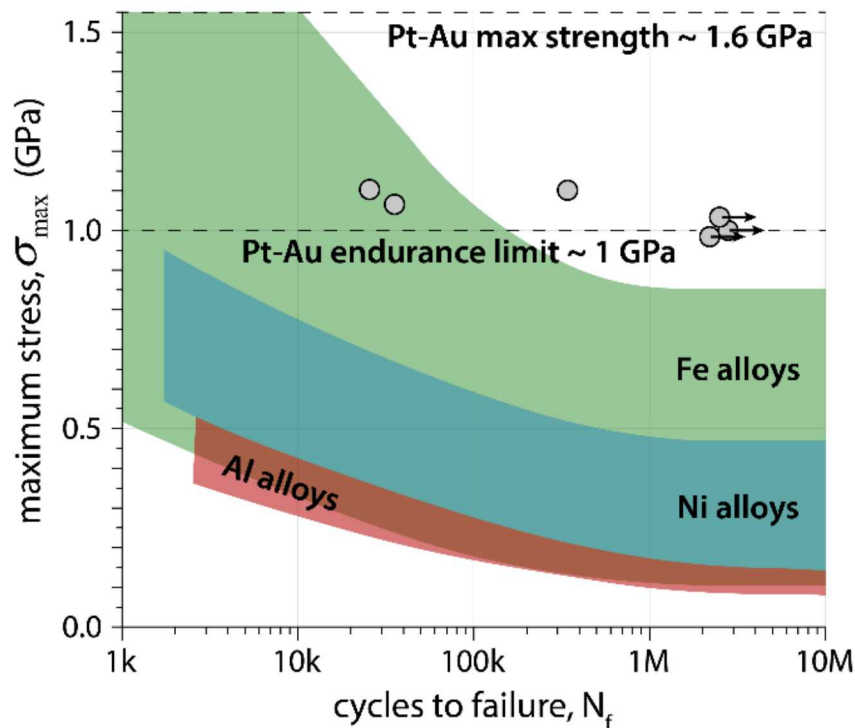


MD Simulation Snapshots (grain growth during uniaxial tensile stress exposure):

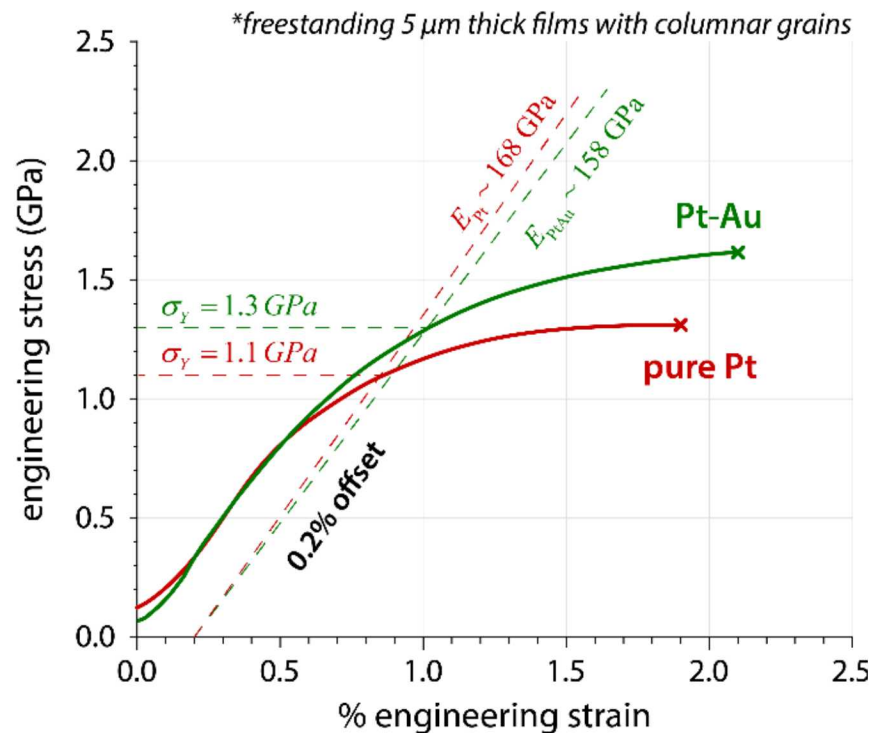


PtAu alloys have high yield strength and remarkable fatigue resistance

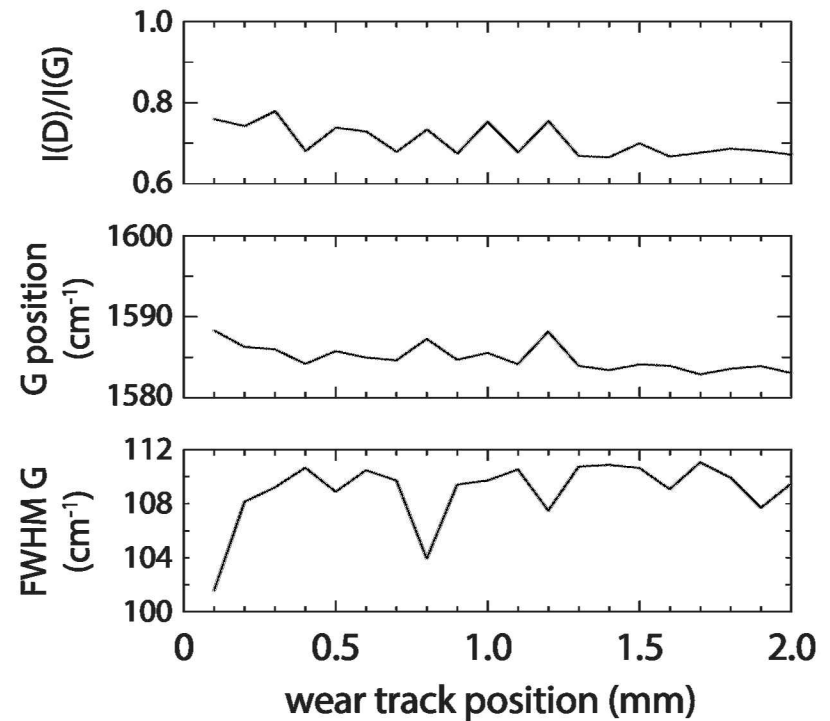
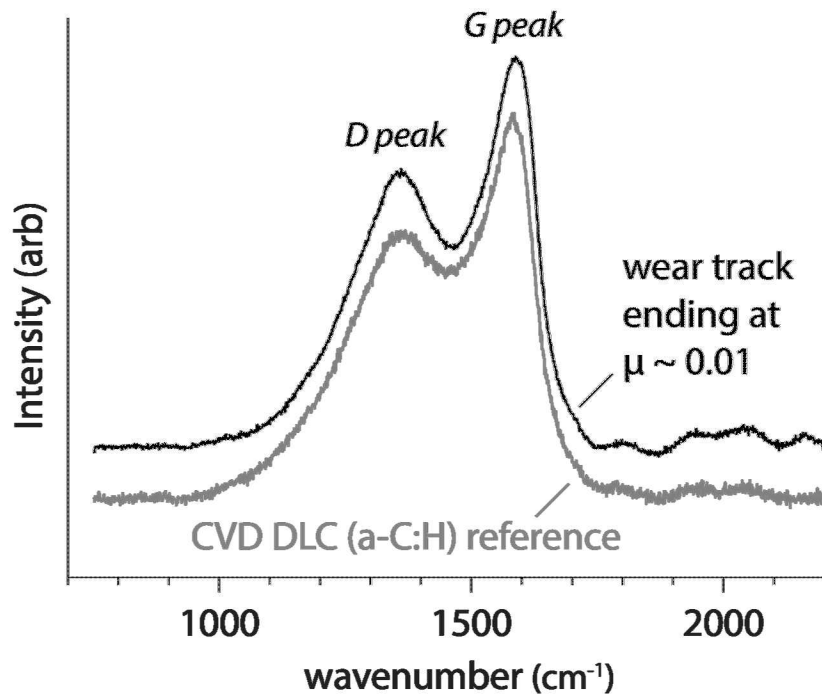
Experimental Tensile Fatigue:



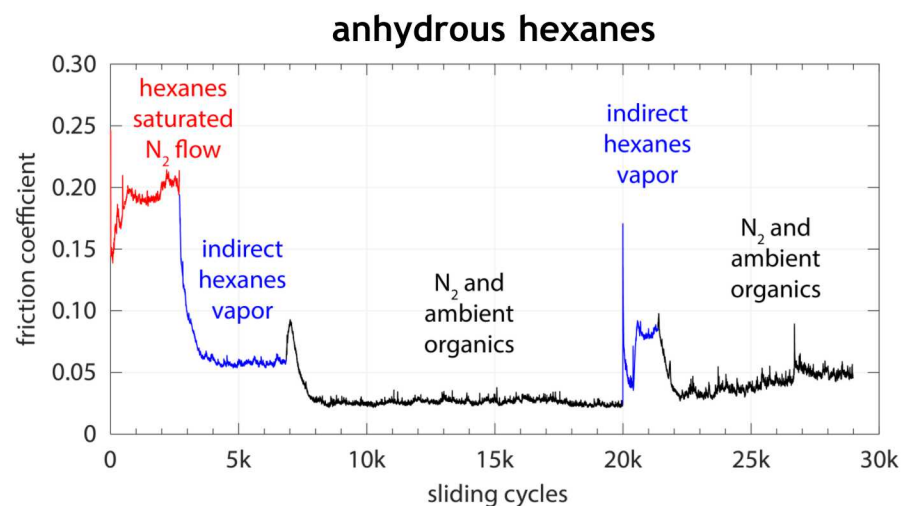
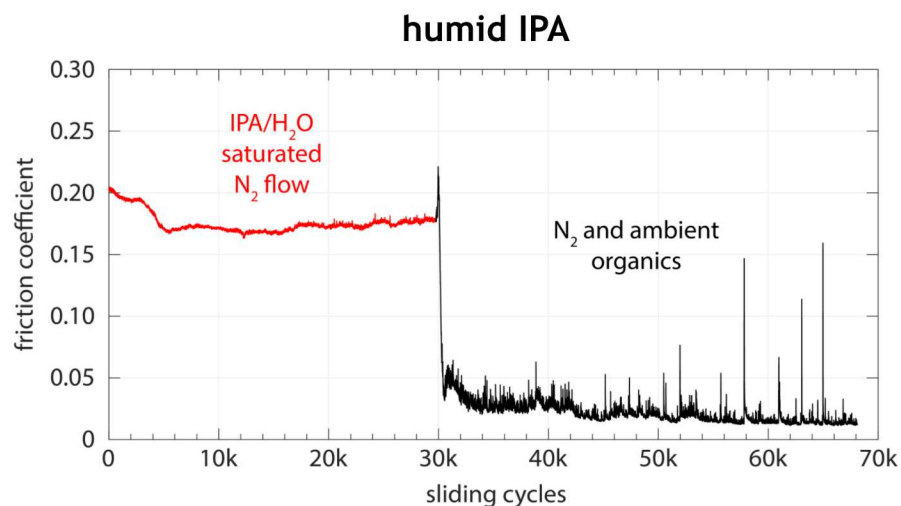
Experimental stress-strain data:



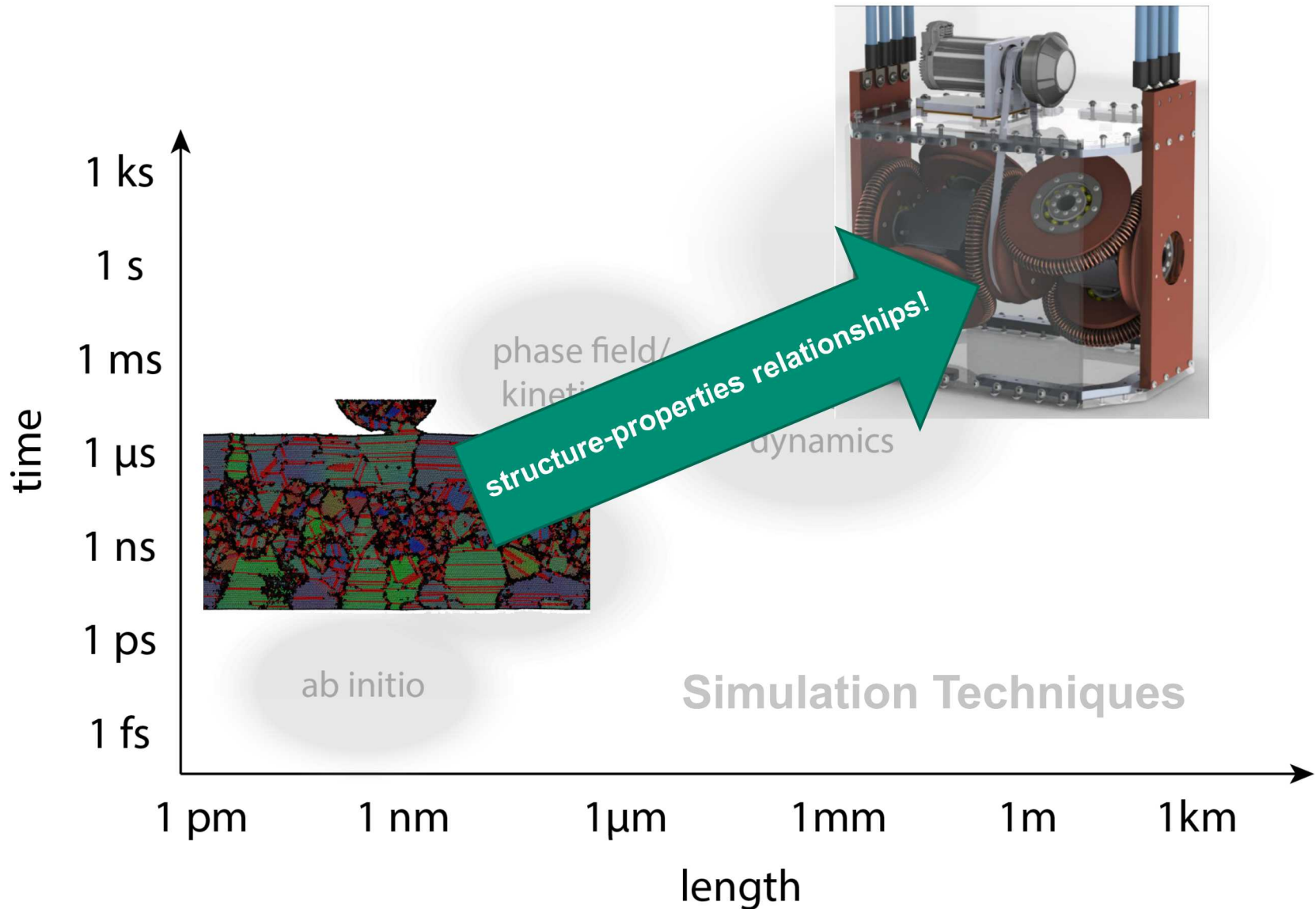
This high yield strength and fatigue resistance generated an interesting result...



elastic recoil showed $\sim 20\%$ H

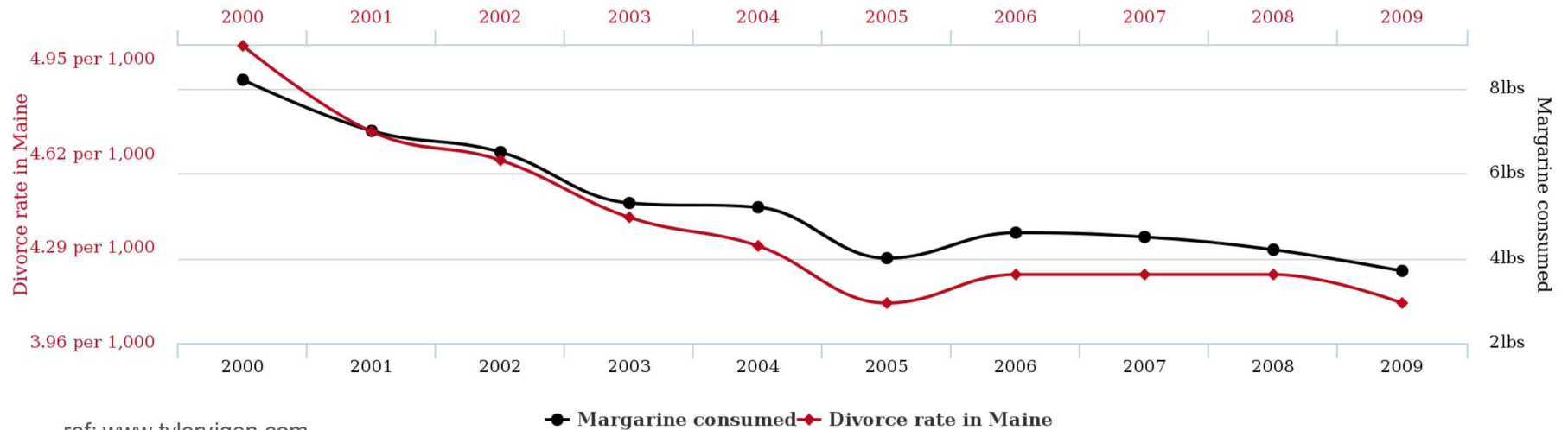


Alcohol helps, but only in moderation.



... as we know, correlation is not causation...

Divorce rate in Maine
correlates with
Per capita consumption of margarine



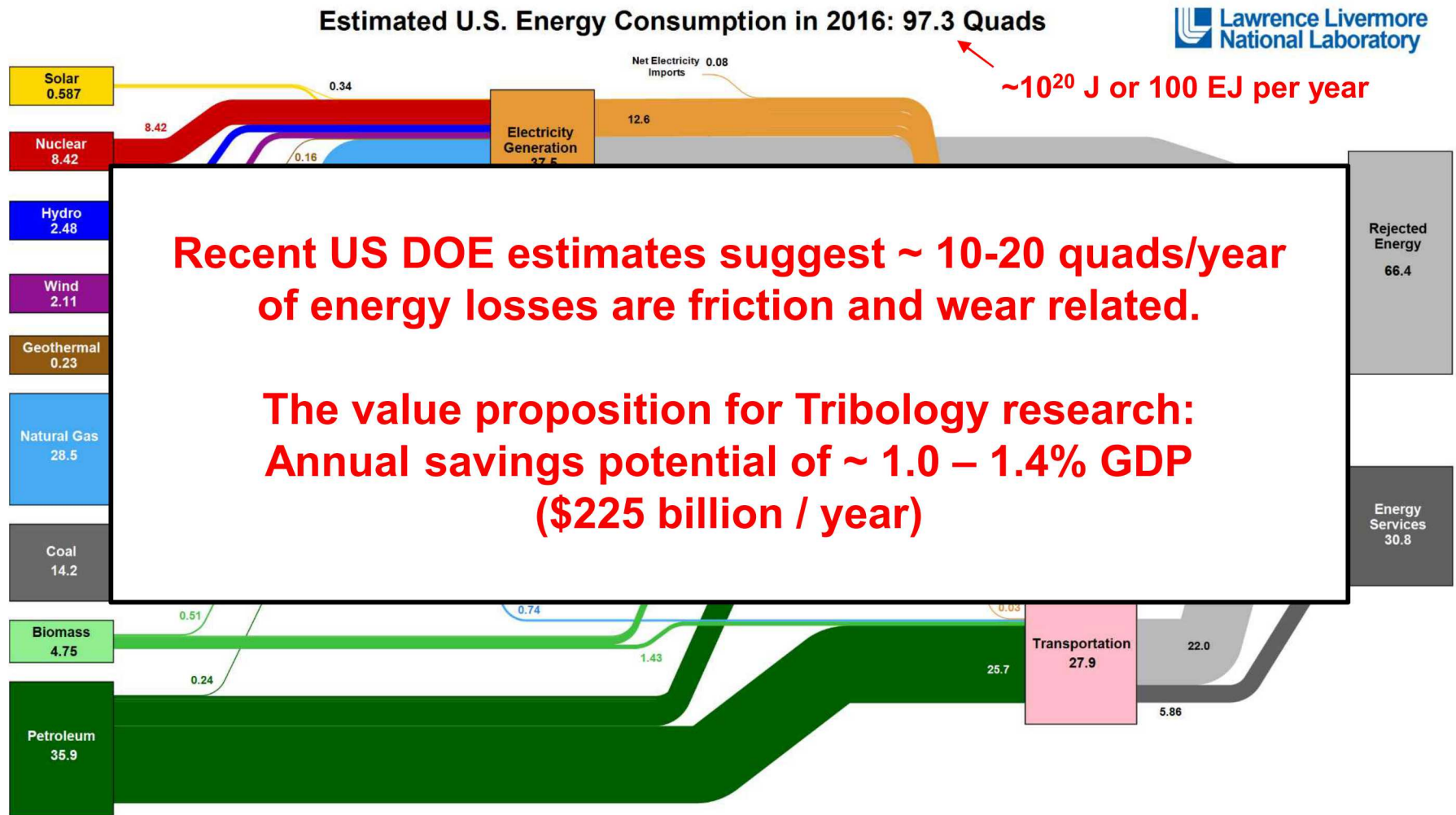
tylervigen.com

so please, for the sake of
married people in Maine...



... use only the pure
stuff

Some Perspective on the Value Proposition of Tribology Research



Source: LLNL March, 2017. Data is based on DOE/EIA MER (2016). If this information or a reproduction of it is used, credit must be given to the Lawrence Livermore National Laboratory and the Department of Energy, under whose auspices the work was performed. This chart was revised in 2017 to reflect changes made in mid-2016 to the Energy Information Administration's analysis methodology and reporting. The efficiency of electricity production is calculated as the total retail electricity delivered divided by the primary energy input into electricity generation. End use efficiency is estimated as 65% for the residential sector, 65% for the commercial sector, 21% for the transportation sector, and 49% for the industrial sector which was updated in 2017 to reflect DOE's analysis of manufacturing. Totals may not equal sum of components due to independent rounding. LLNL-MI-410527

Attempts to define wear & friction regimes remain empirical/phenomenological

Scripta METALLURGICA
et MATERIALIA

Vol. 24, pp. 805-810, 1990
Printed in the U.S.A.

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VIEWPOINT SET No. 14

WEAR-MECHANISM MAPS

M. F. Ashby* and S. C. Lim*,

*Engineering Department, Cambridge Univer
+National University of Singapore, Kent R

(Received August 15
(Revised October 16

WEAR-MECHANISM MAPPING:

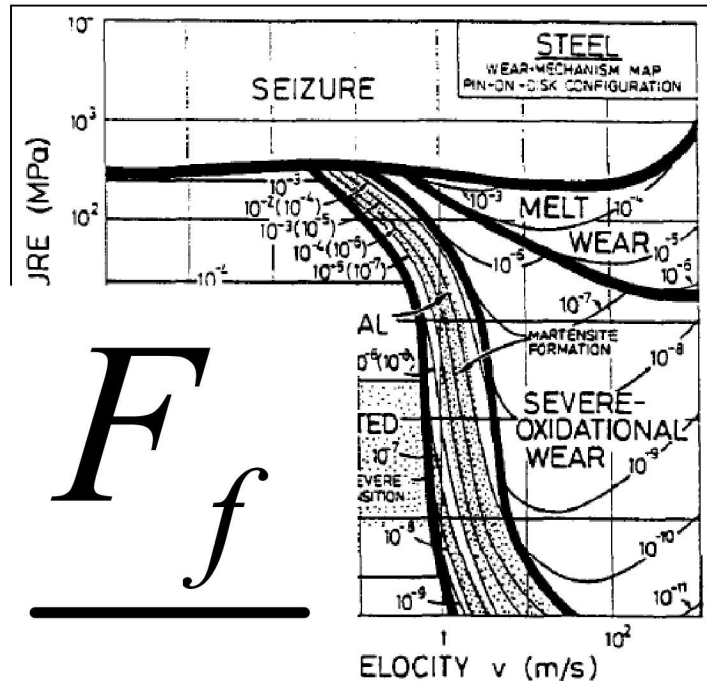
Wear is the loss or transfer of material. In general, the wear rate W (defined here as the volume of material lost per unit distance slid) depends on the bearing pressure p carried by the contact and A_n is its nominal area and on the material properties and geometry of

$$W = f(F/A_n, v, \text{Mat. Props.}, \text{Geomet})$$

But one such equation is not enough. There are many mechanisms of wear, each dependent in a different way on the variables. F and v , is the one leading to the fastest rate of wear. Mechanisms encountered in wear studies of metal melting, by chemical change induced by friction, plasticity and by brittle fracture.

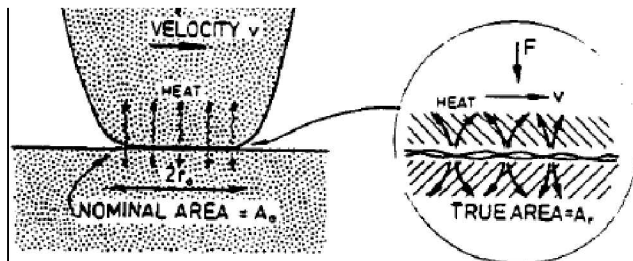
TABLE 1: MECHANISMS OF WEAR

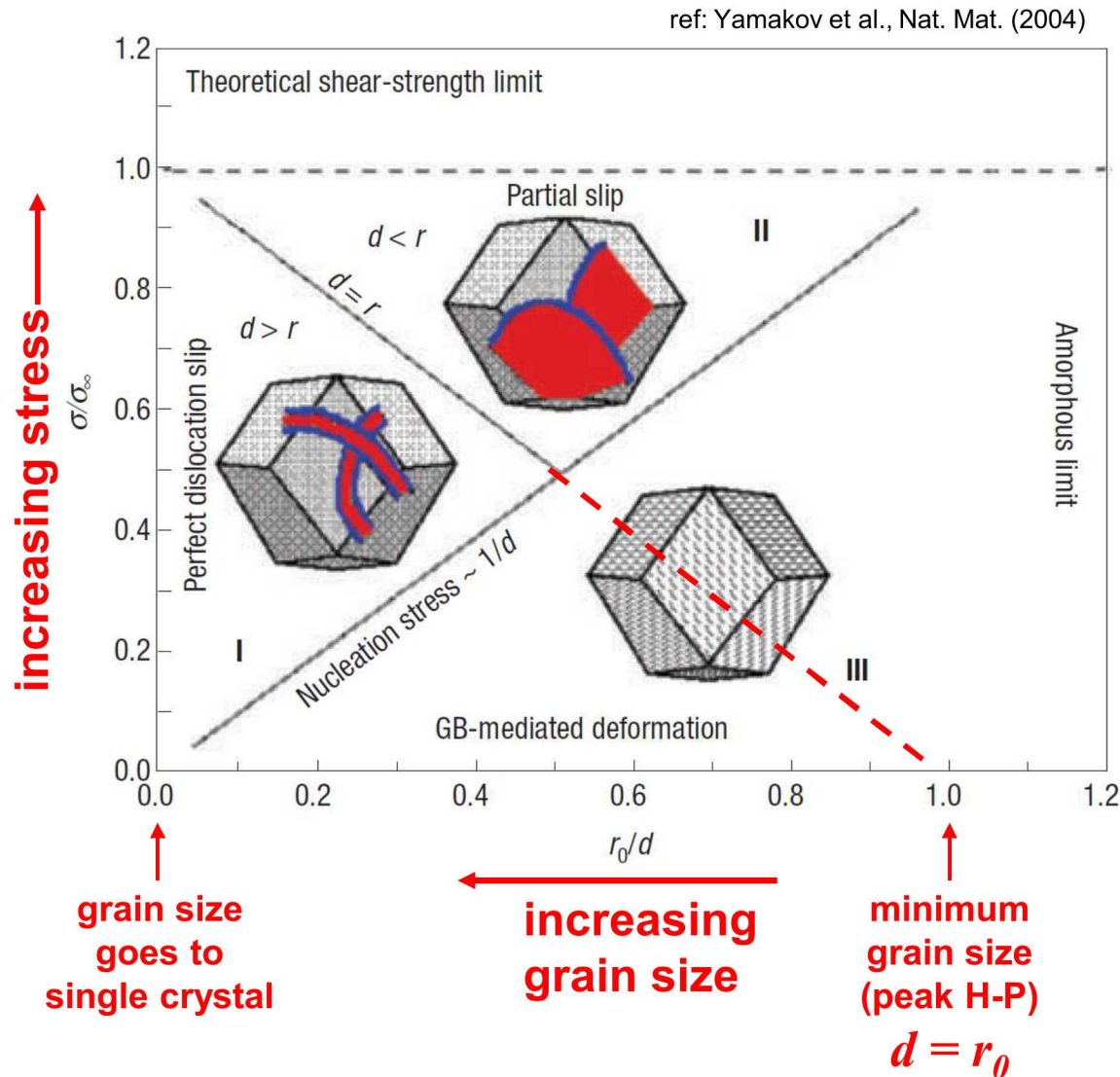
| METALS | CERAMICS |
|---------------------------|------------------------|
| SEIZURE | SEIZURE |
| MELT WEAR | MELT WEAR |
| SEVERE-OXIDATIONAL WEAR | THERMALLY INDUCED WEAR |
| MILD-OXIDATIONAL WEAR | THERMAL CRACKING |
| PLASTICITY-DOMINATED WEAR | BRITTLE FRACTURE |
| ULTRA MILD WEAR | TRIBOCHEMISTRY |



mechanism map for low-
on physical modelling
periments. The shaded
regions.

$$\mu = \frac{F_f}{F_n}$$





**Equilibrium (zero stress)
dislocation splitting distance:**

$$r_0 = \frac{(2 + \nu) G b^2}{4\pi (1 - \nu) \gamma_{sf}}$$

**Stress-dependent splitting
distance:**

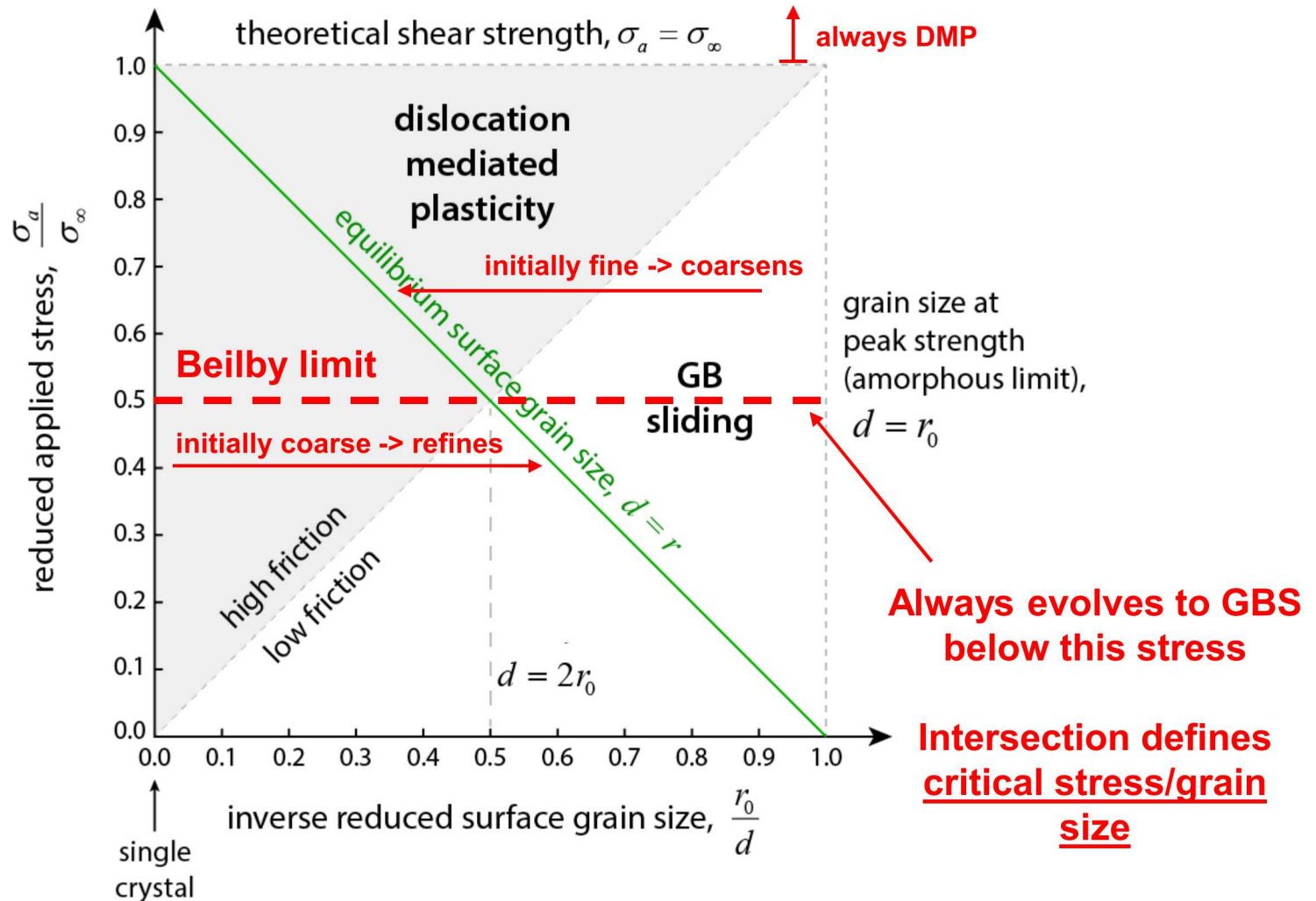
$$r = \frac{r_0}{1 - \sigma_a / \sigma_\infty}$$

Theoretical shear strength:

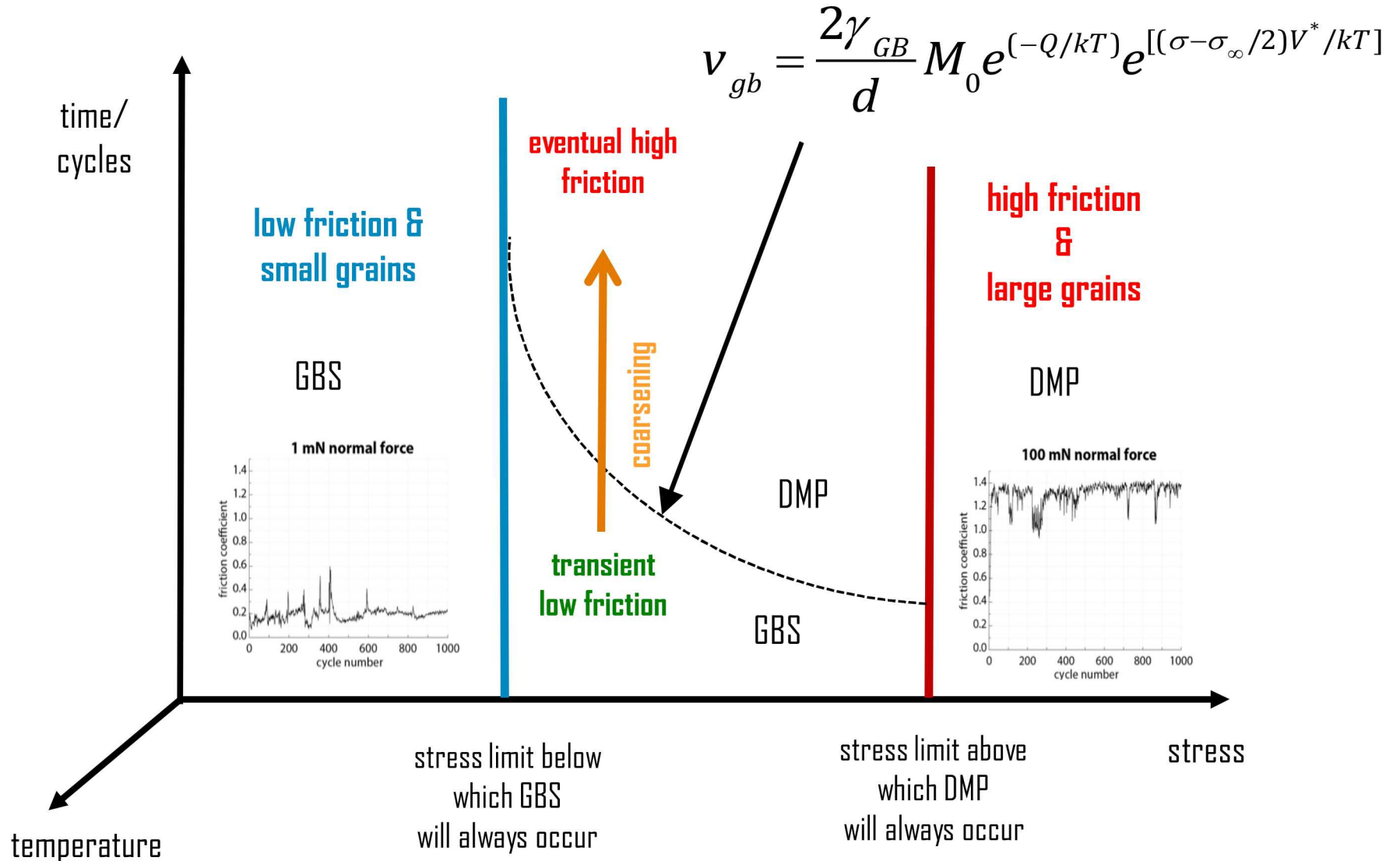
$$\sigma_\infty = \frac{2\gamma_{sf}}{b}$$

Ref: Froseth et al., Acta Mat. (2004)

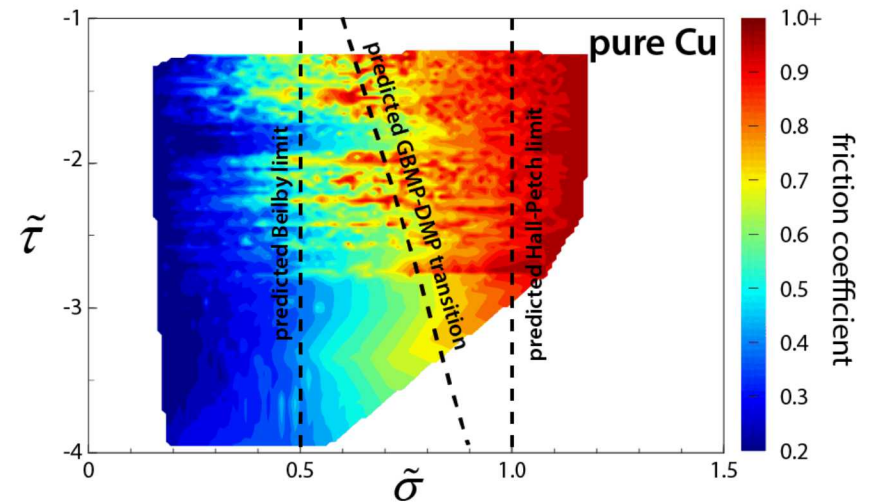
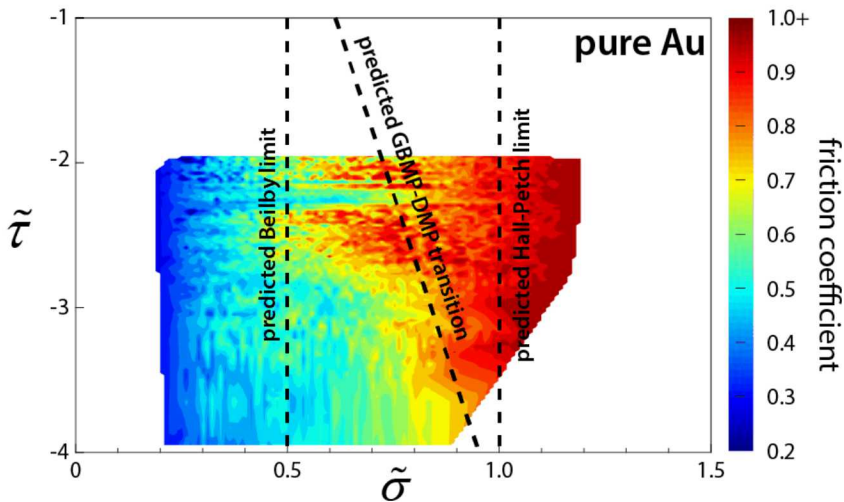
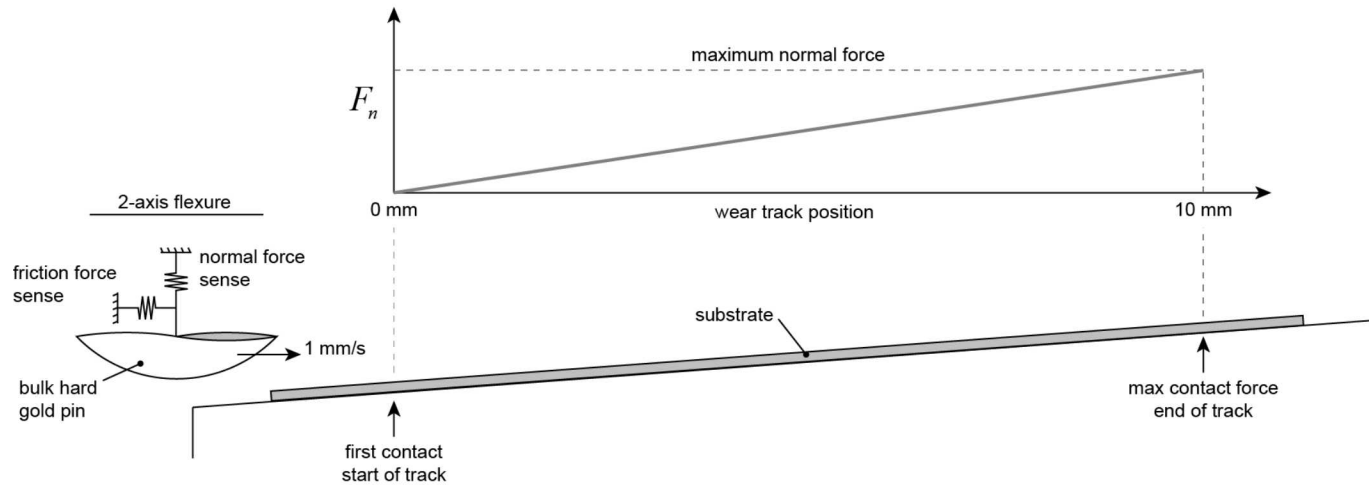
Recasting the Yamakov et al. criteria for tribology (& time-dep)



Generalized friction regimes map for metals



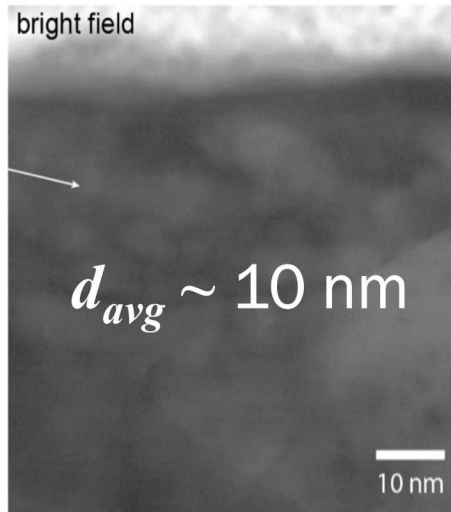
How can we test theory in one shot? Ramped contact force experiments



We can predict a crossover critical grain size

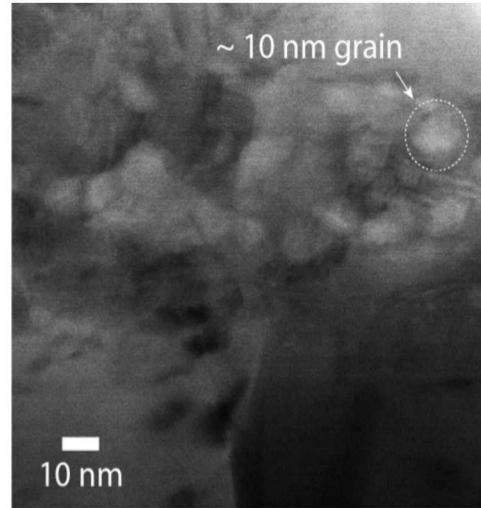
gold on gold

low friction



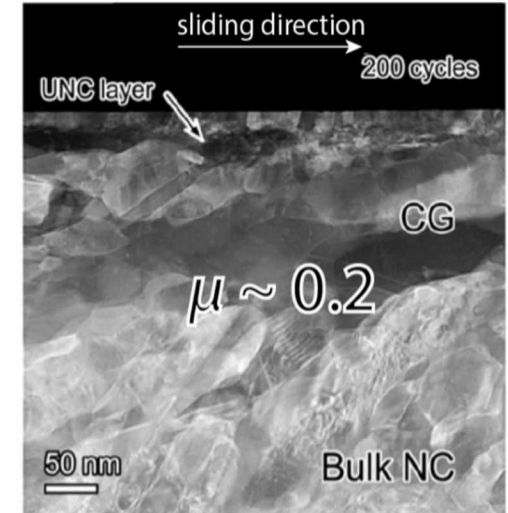
critical grain size = 17.4 nm

copper on copper



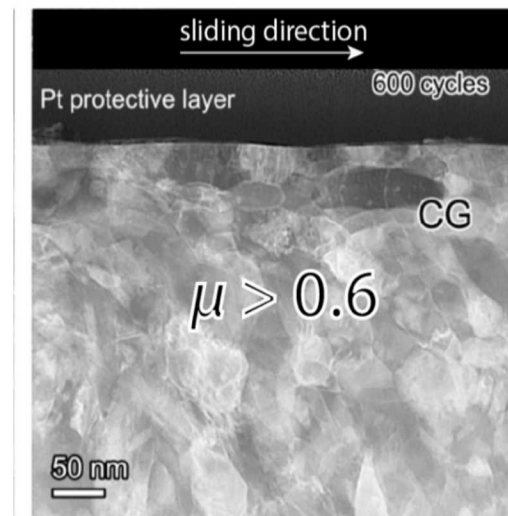
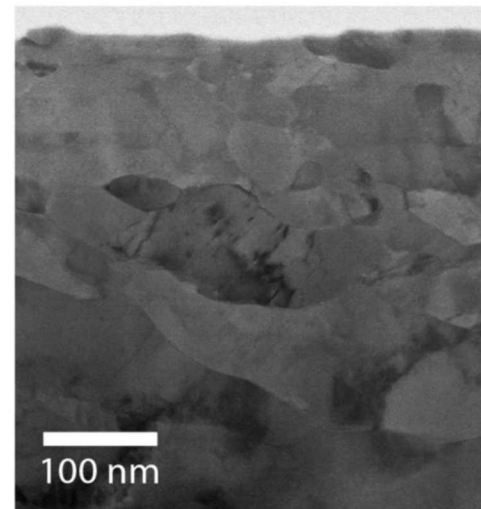
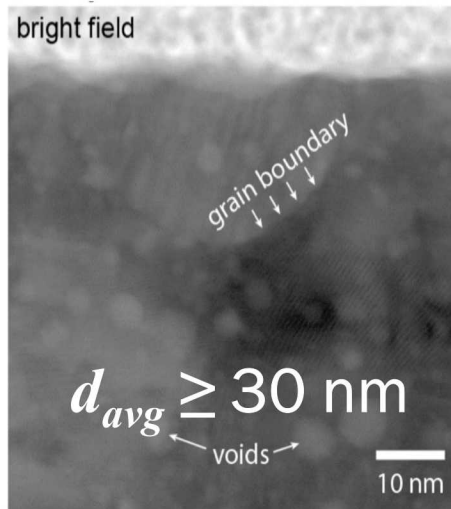
critical grain size = 11.8 nm

Si_3N_4 on nickel

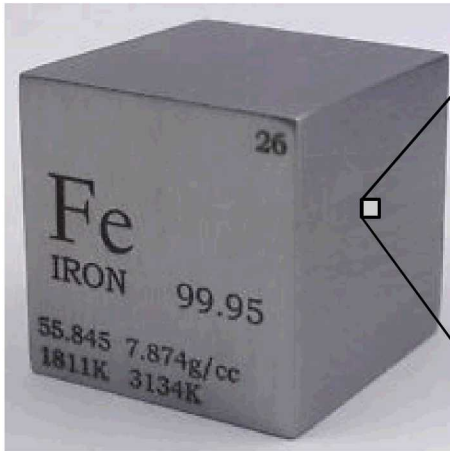


critical grain size = 9.8 nm

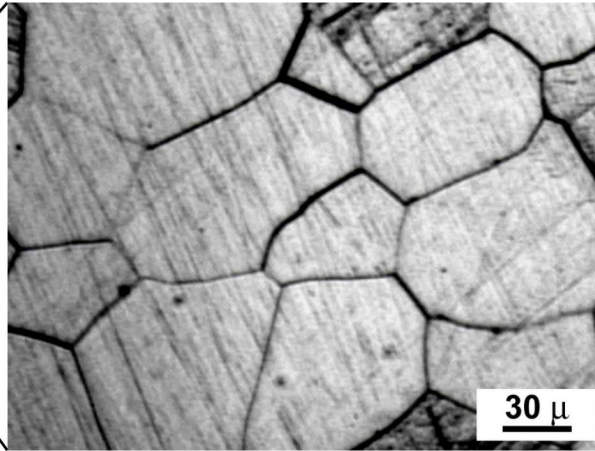
high friction



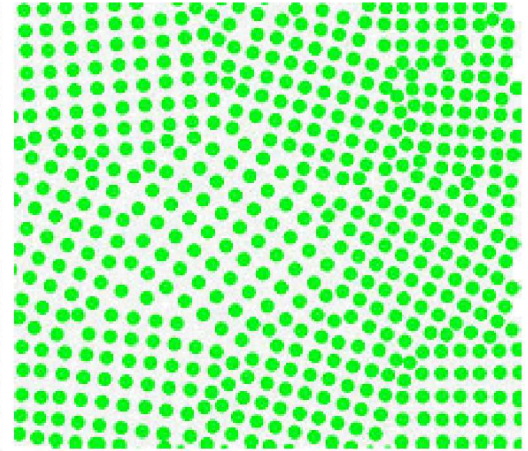
Crystal Structure of Metals (A Primer)



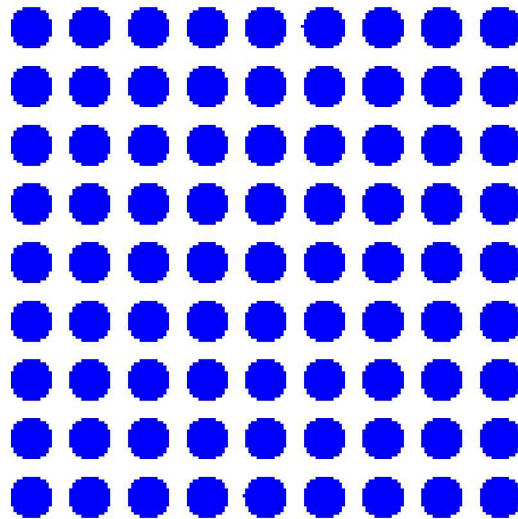
<https://www.amazon.com/25-4mm-Metal-99-95-Engraved-Periodic/dp/B06WVQ7JXD>



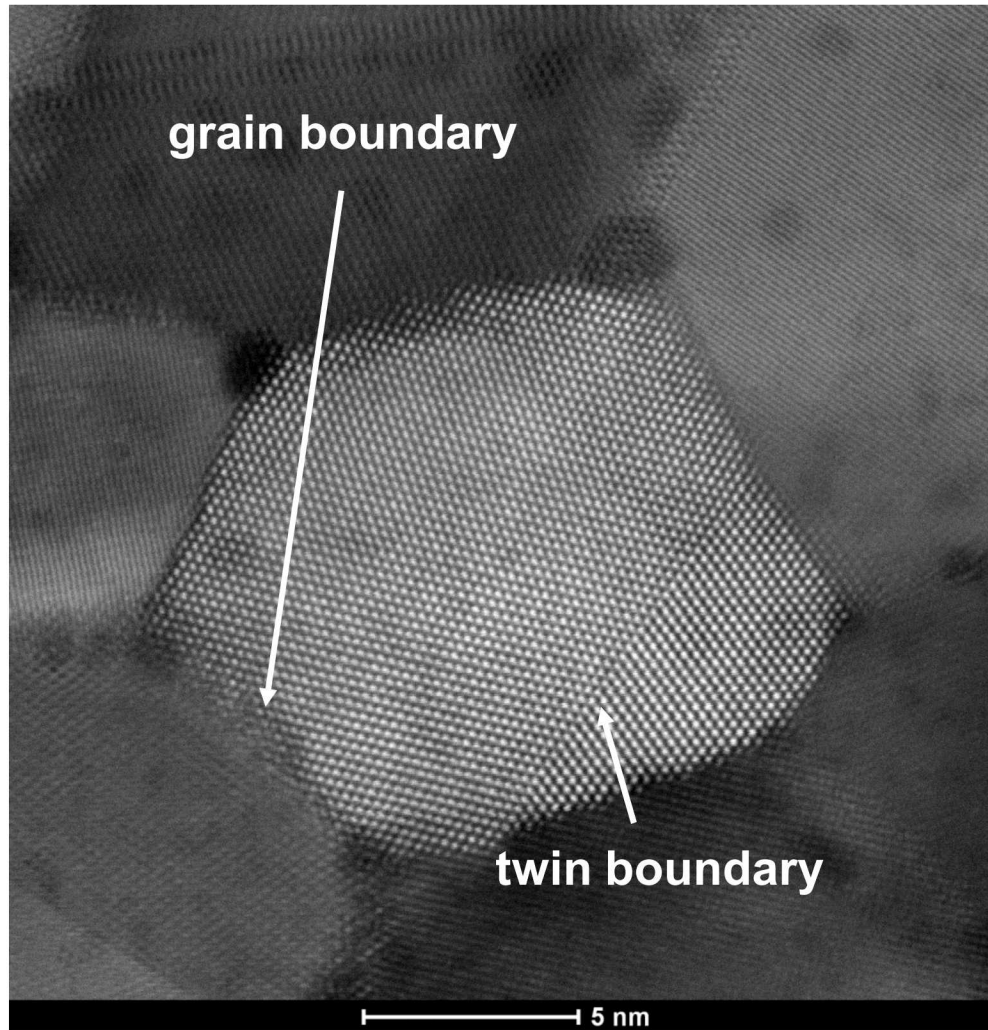
https://en.wikipedia.org/wiki/Grain_boundary



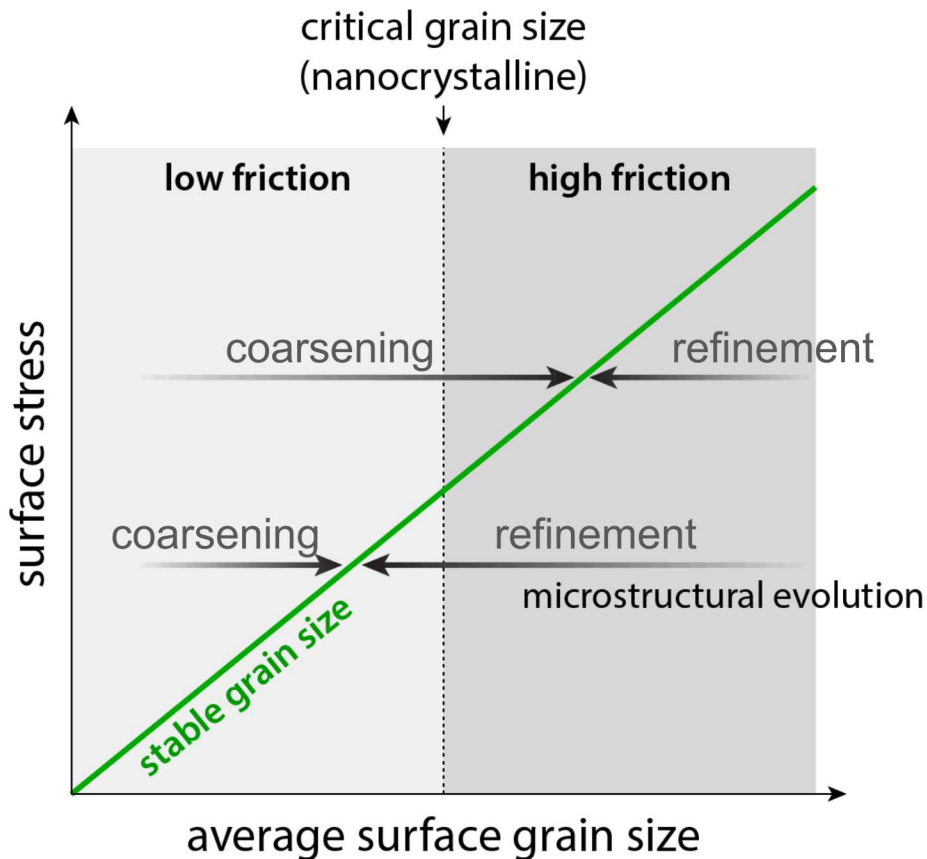
Shear via
dislocation
motion:



*"Crystals are like people, it's the defects
that make them interesting."
- C. J. Humphreys*



Grain size evolution is a competition (growth/refinement)



“All models are wrong, but some are useful.”
--George Box

“It is better to be vaguely right than exactly wrong.”
--Carveth Read

- Effective refinement from recrystallization (Zener & Holloman, 1944; McQueen et al., 1967)
- Known in rocks and ice cores (Derby et al., 1992)
- Recently extended to metals under severe plastic deformation (Pougis et al., 2014)