

New Insights into Alloy Design for Tribological Applications



PRESENTED BY

Dr. Nicolas Argibay



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Motivation for Metals Tribology Research

wind turbine slip-rings
(sensors and blade pitch motors)



**Estimated 150 Metric Tons (\$6.9B) of Au
used in Electrical Contacts per Year:**

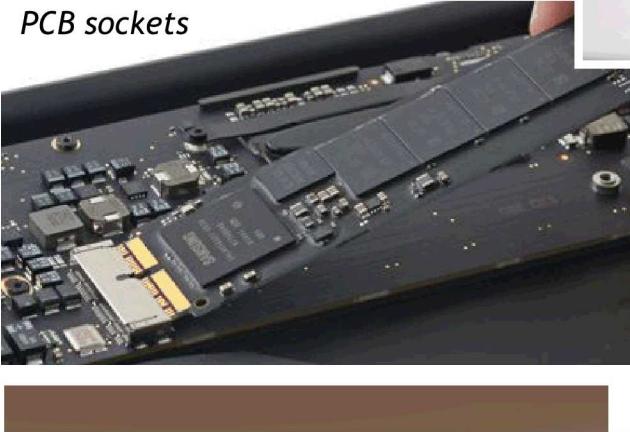
Refs: Gold Survey, Gold Fields Mineral Survey Ltd, 2011
Gold Bulletin 2010, Vol. 43-3, C. Hagelüken and C.W. Corti,
Gold Bulletin 1986, Vol. 19-3, T.D. Cooke

UEA Inc.

EV charging



PCB sockets



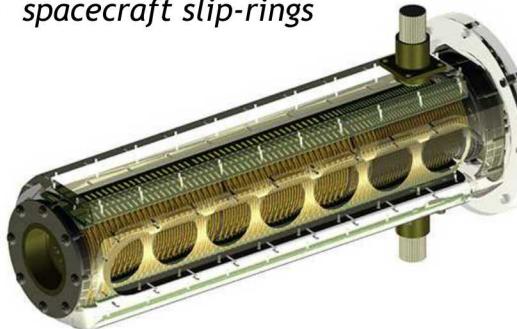
CPU sockets



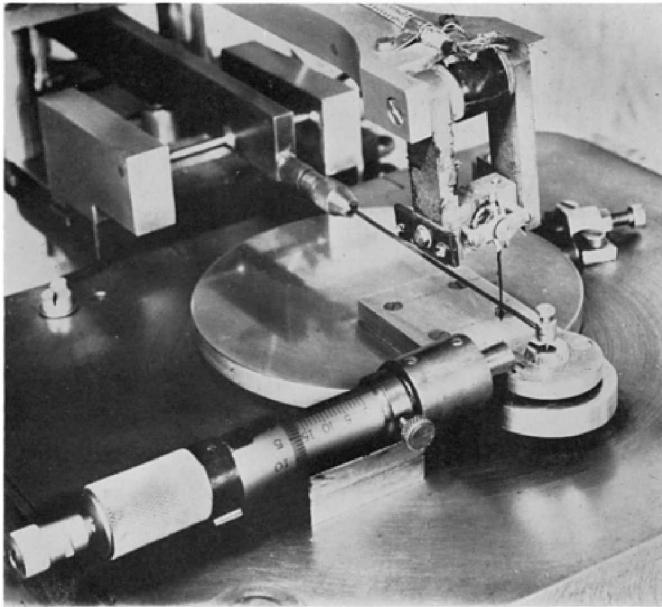
45 connectors



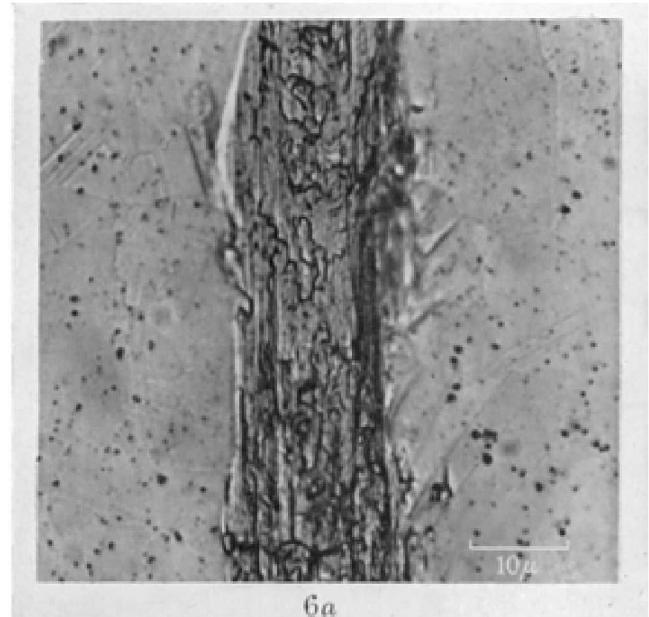
spacecraft slip-rings



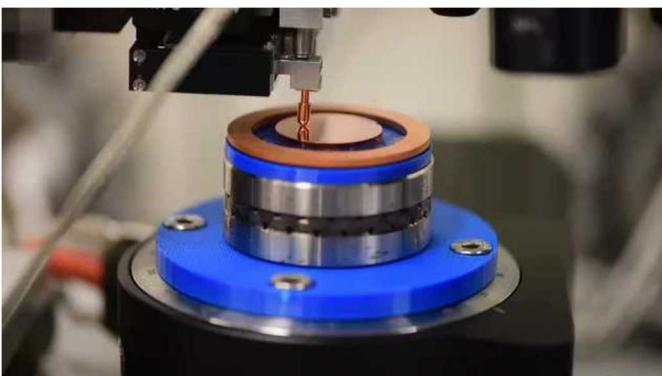
Metals Often Have Poor Tribological Properties



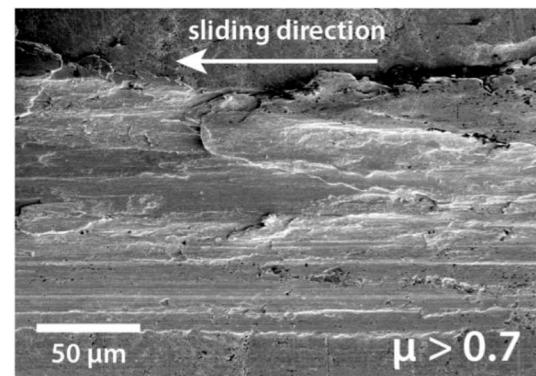
Whitehead, Proc. Royal Soc, 1950



6a
Cu in air @ 150 mN, $\mu > 1.6$

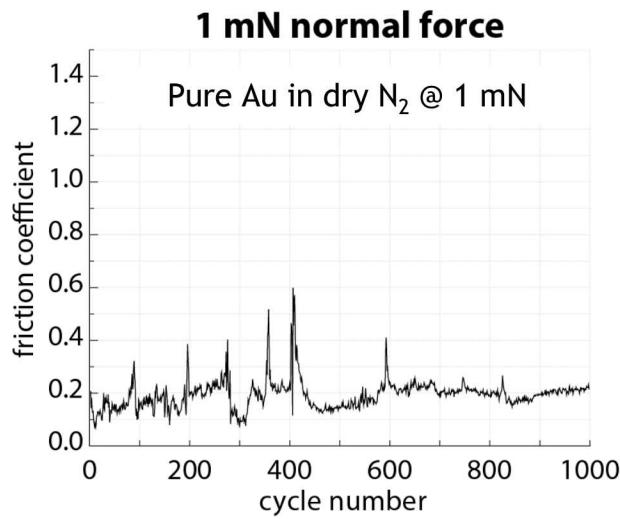


Metals tend to show strong, rapid interfacial bonding (cold welding) with galling.

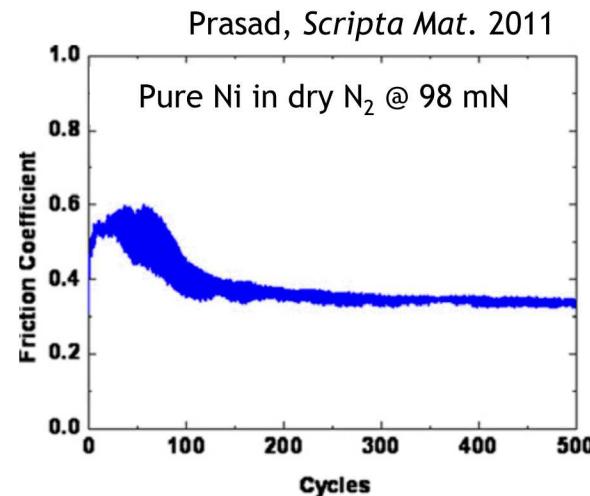


Au in dry N_2 @ 75 mN, $\mu > 0.7$

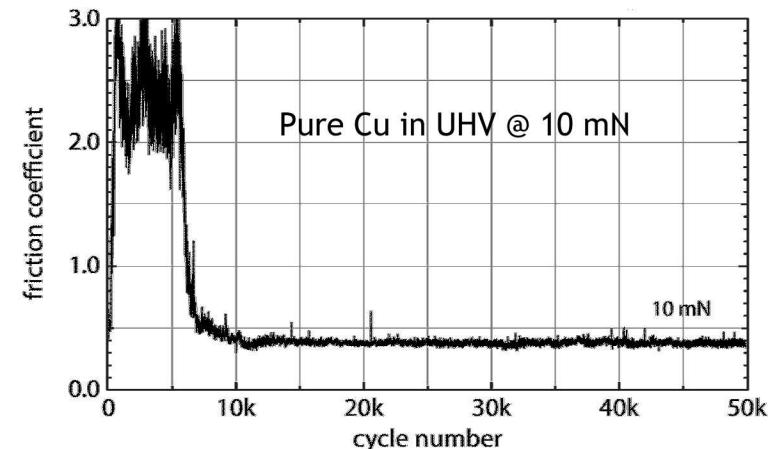
More recent results: Low Friction starting from Bare, Soft, Pure Metals



"When surfaces are cleaned in a good vacuum, the sliding friction... becomes vanishingly small."
- Bowden & Hughes, *Nature* 1938.



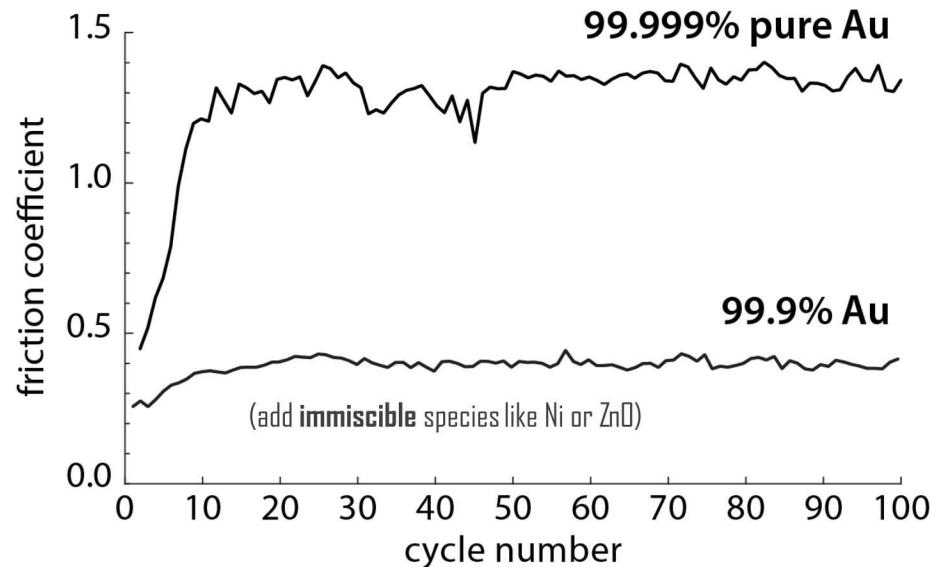
"It was found quite unexpectedly that with some metals, very low friction less than 0.10 was observed."
- Tamai, *J. Appl. Phys.* 1961



Low friction with pure metals is achievable.

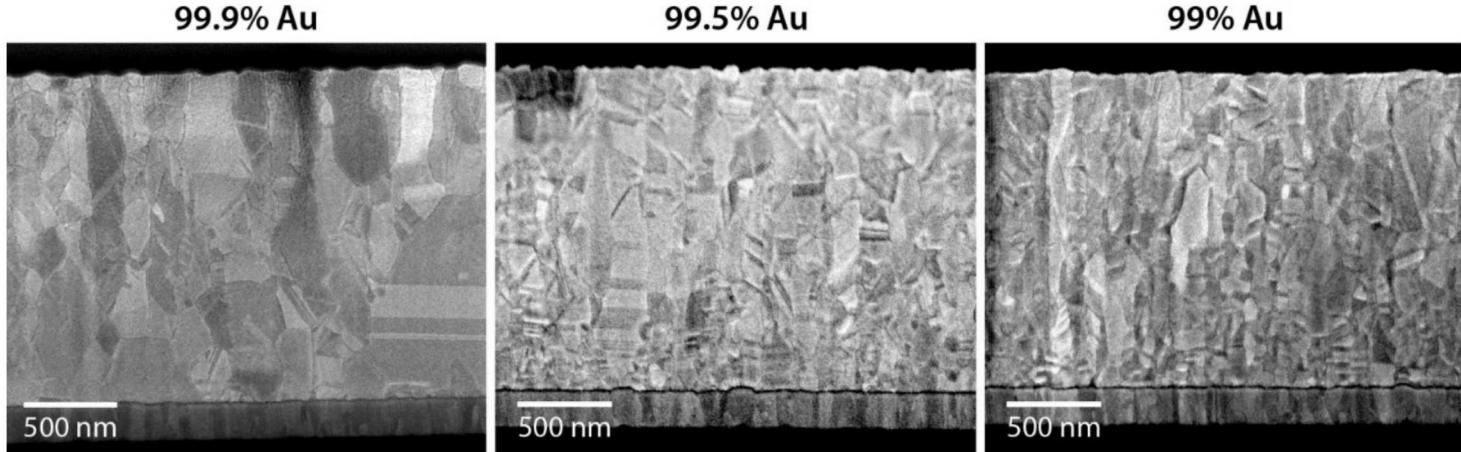
Low Friction and Wear of Nanocrystalline Metals

Alloying reduces friction coefficient:

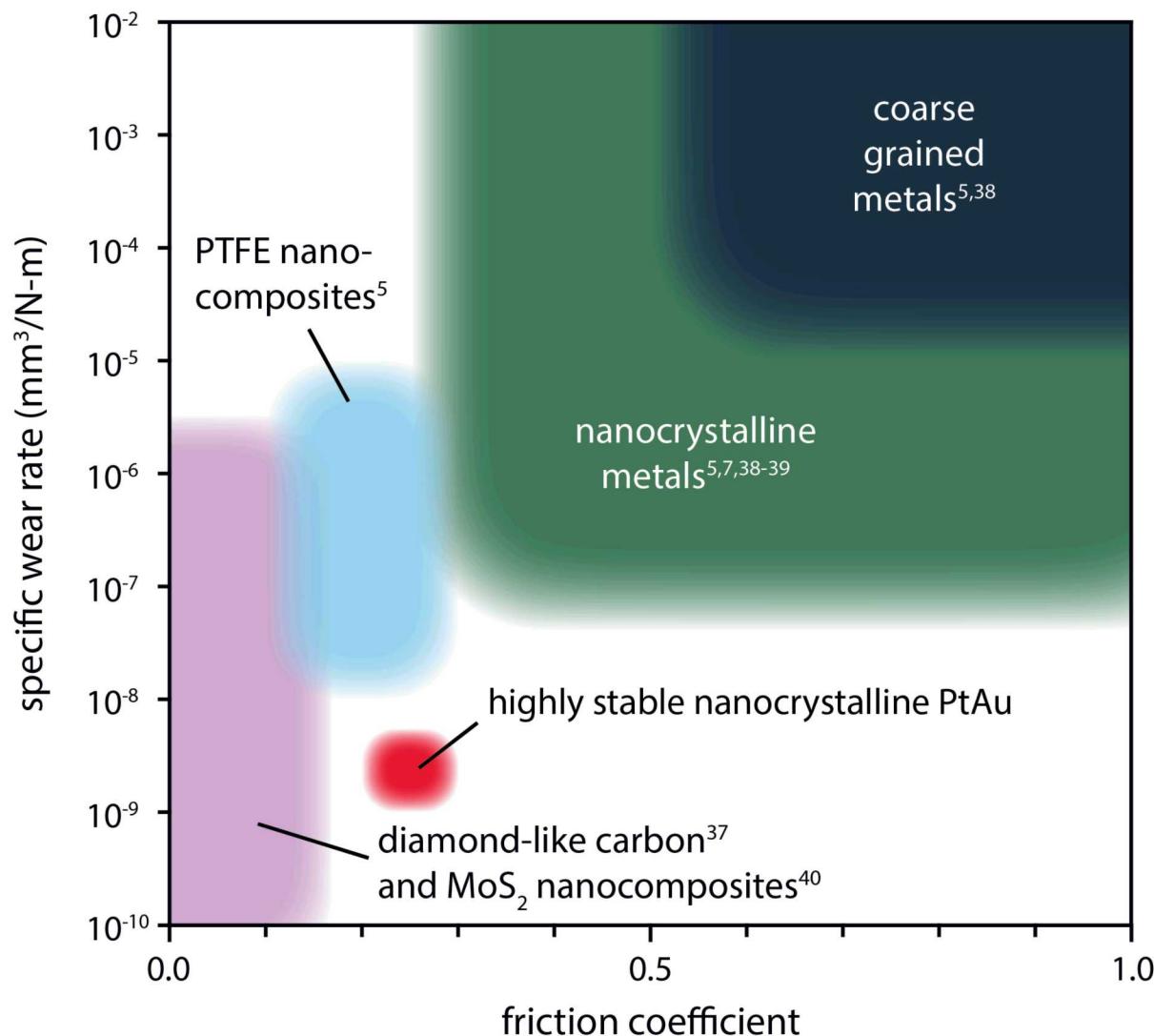


Alloying improves friction & wear performance by reducing and stabilizing grain size

...by reducing grain size:

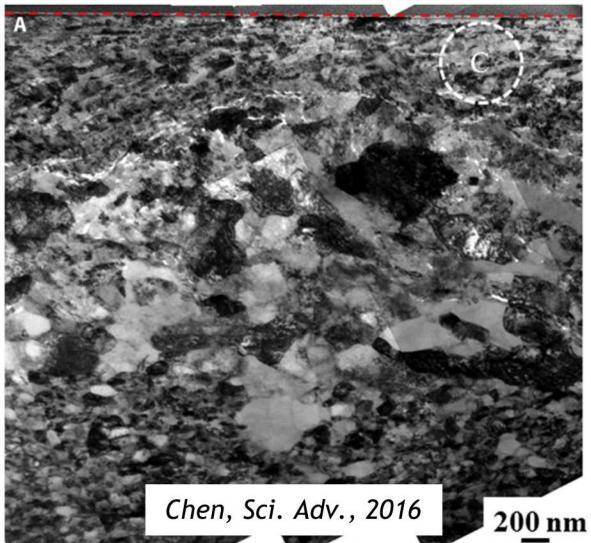


Low Friction and Wear of Nanocrystalline Metals

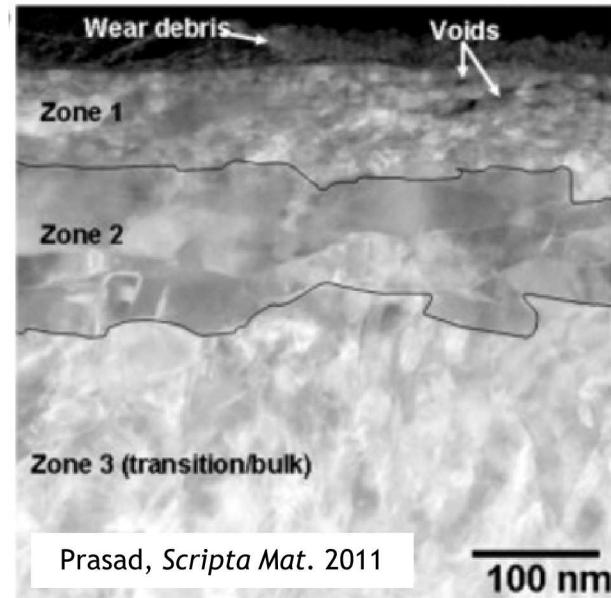


What do we know now?

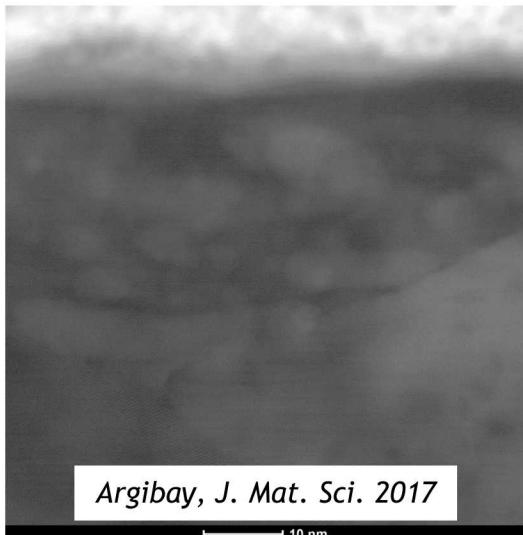
Pure Cu in air @ 50 N, $\mu \sim 0.37$



Pure Ni in dry N₂ @ 98 mN, $\mu \sim 0.2$



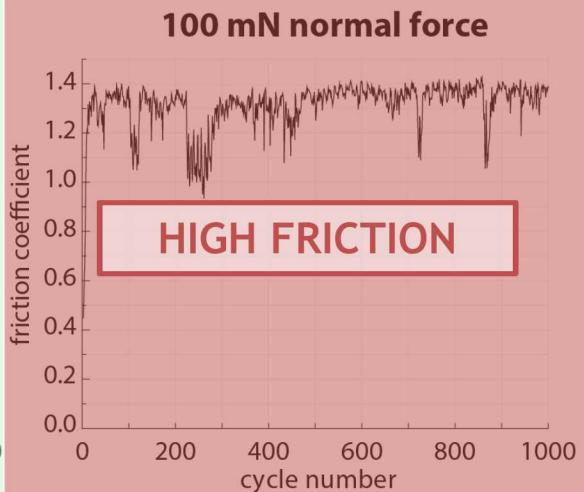
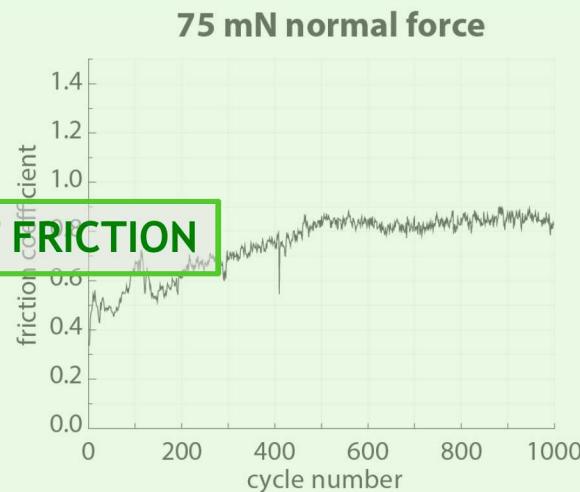
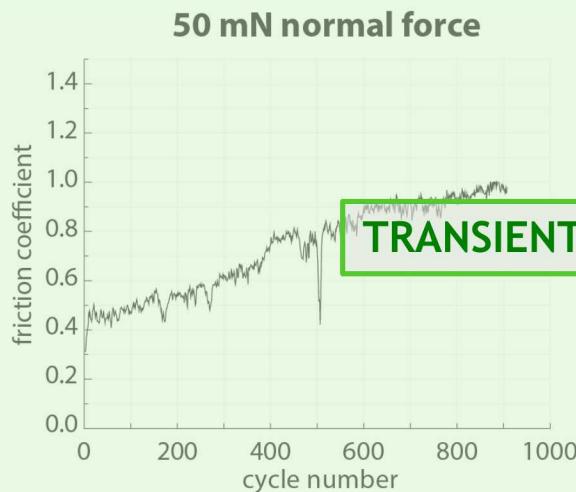
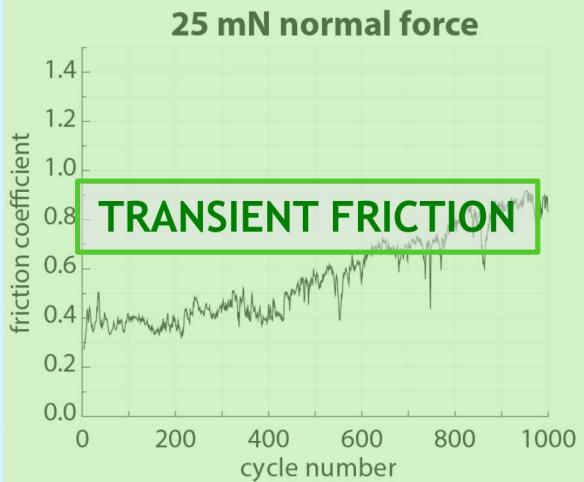
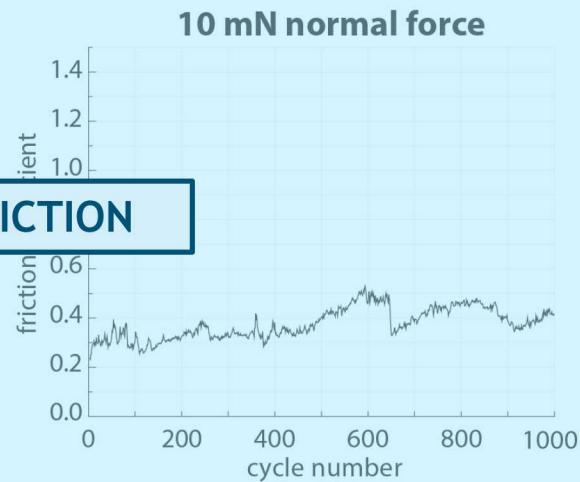
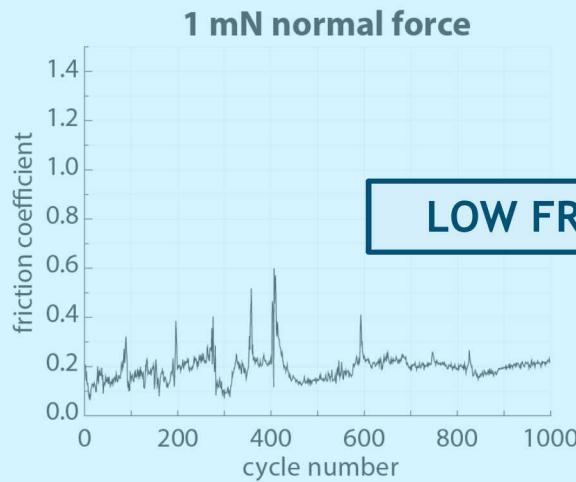
Pure Au in dry N₂ @ 1 mN, $\mu \sim 0.2$



Low friction is associated with the formation of a highly surface localized UNC layer.

What do we know? Friction depends on applied load

(Pure Au in dry N₂)

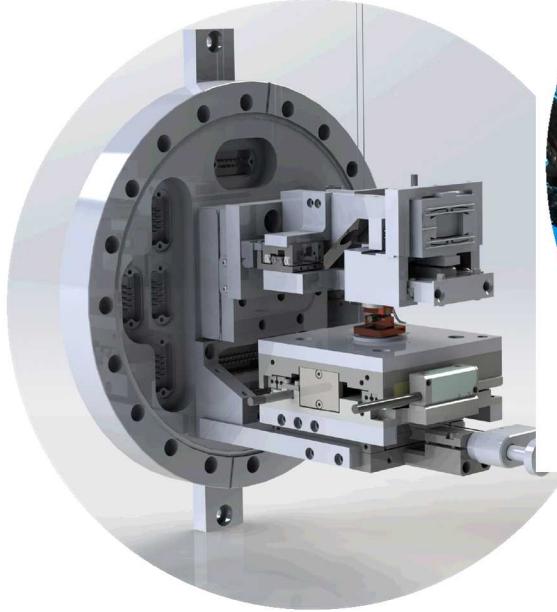


Three different friction regimes, with transitions.

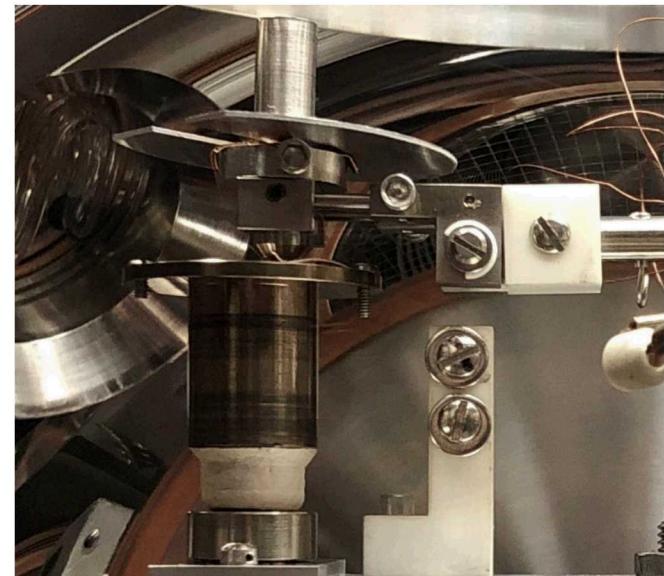
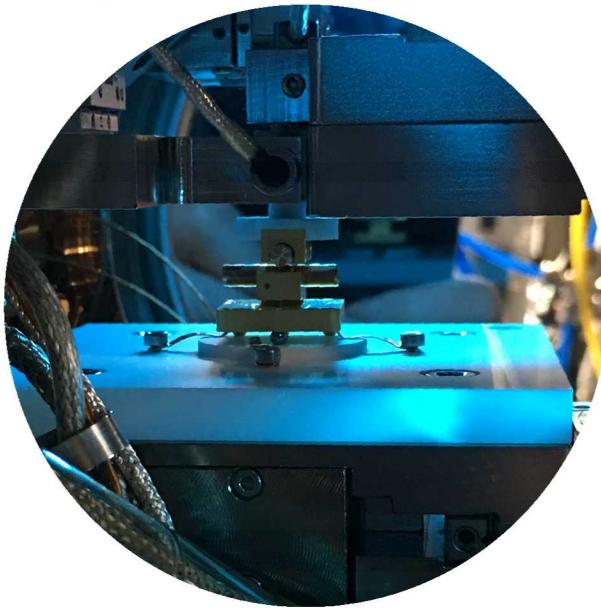
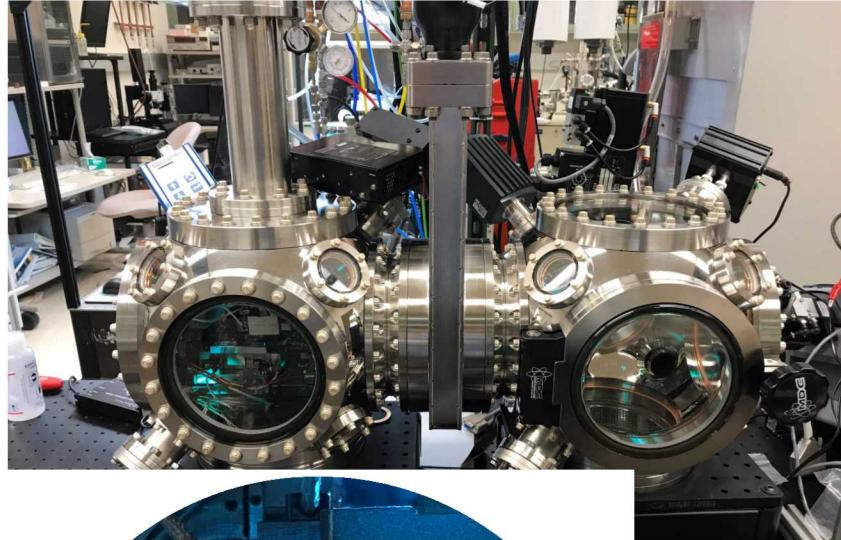
9 | What about adsorbates? Test in UHV

Left Module:

- Linear Reciprocator
- Load metering
- Cryo stage (4-800K)
- 10^0 to 10^{-9} torr
- 0.1 mN to 1 N
- 100 μ m/s - 100 mm/s
- 10 kHz acquisition
- capacitive displacement sensors & flexural cantilevers



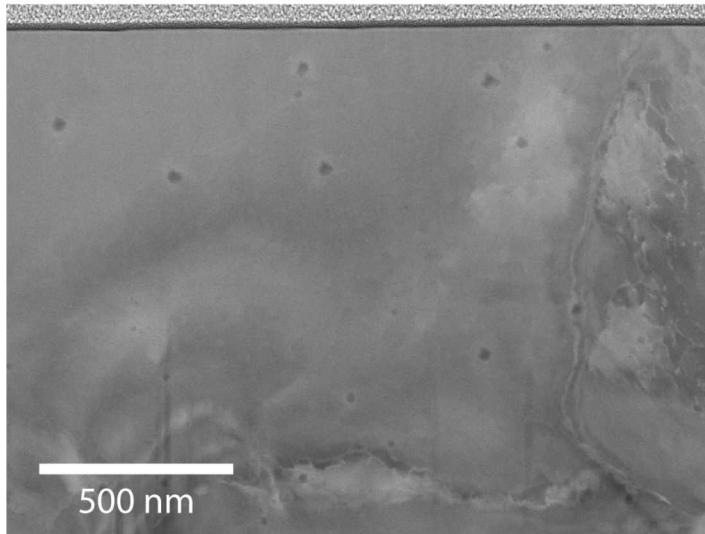
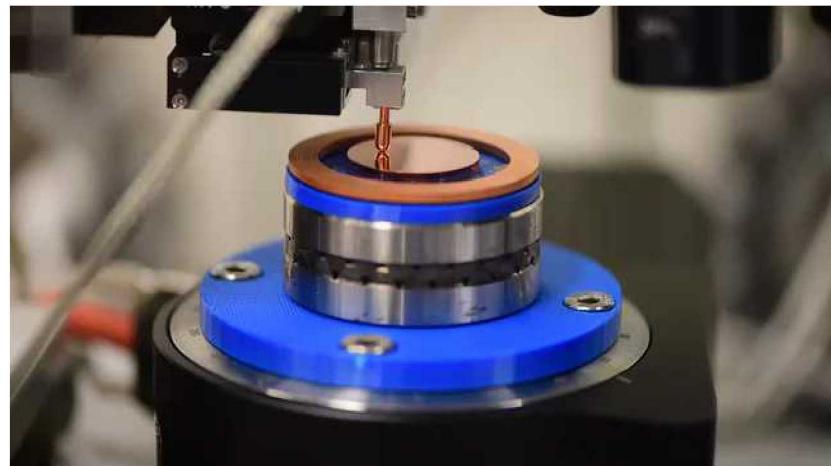
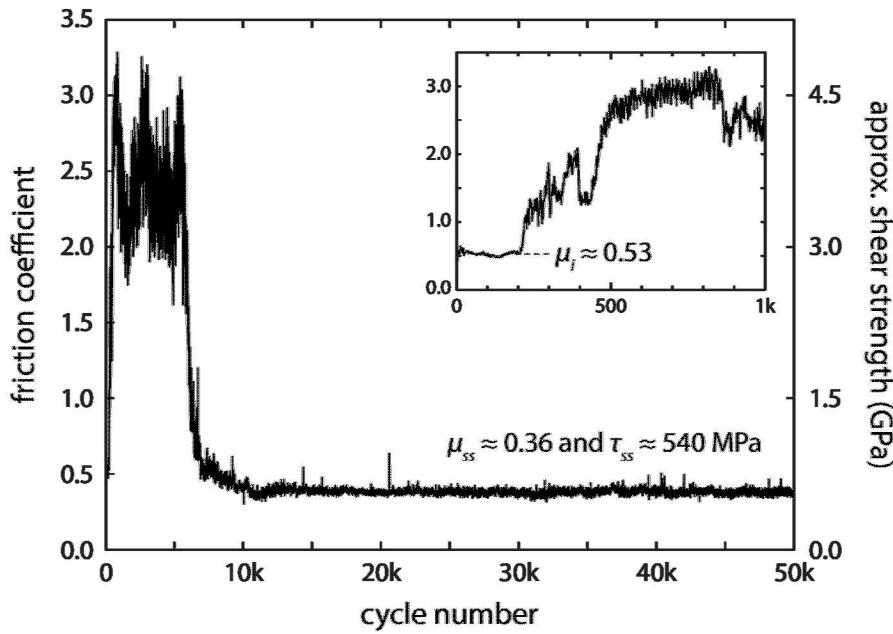
*crossed-cylinders
and sphere-on-flat
configurations*



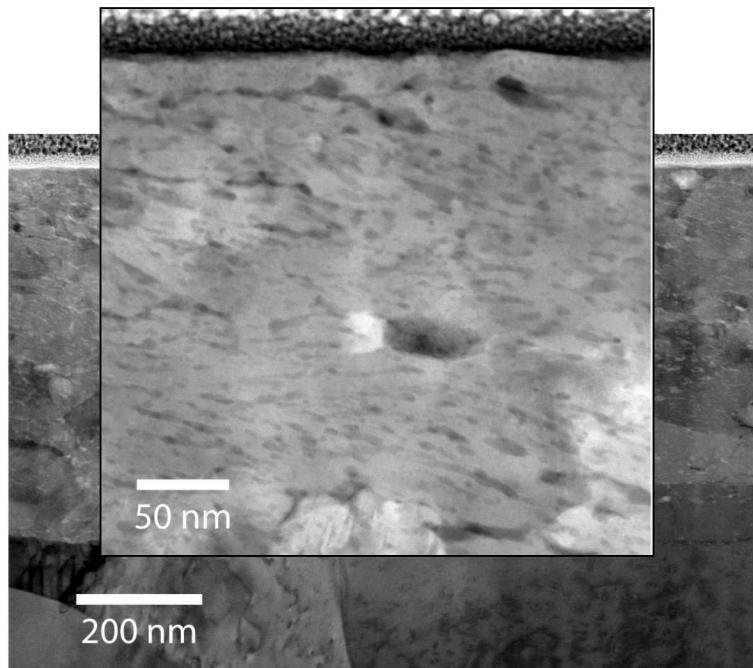
Right Module:

- Rotary Module
- Dead weight
- 5 mN to 10 N
- 100 Hz acquisition
- 10^0 to 10^{-9} torr
- Strain gage sensor

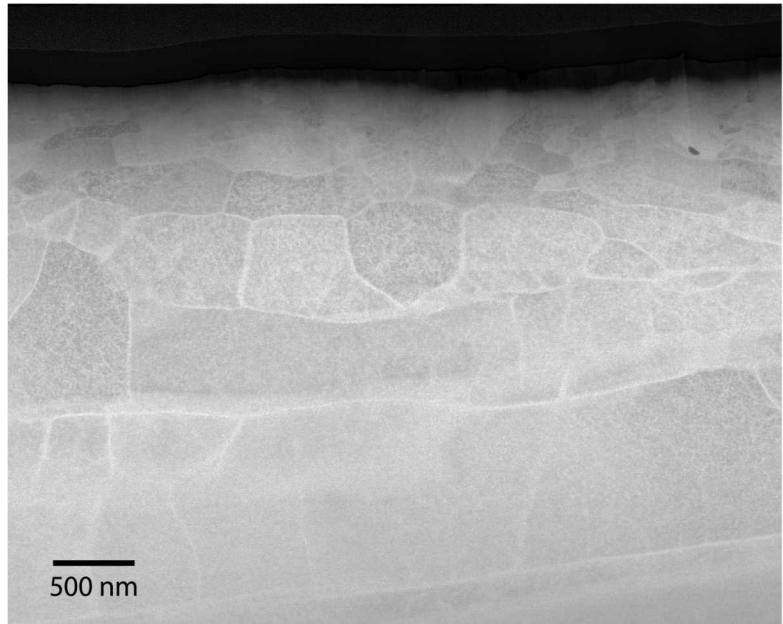
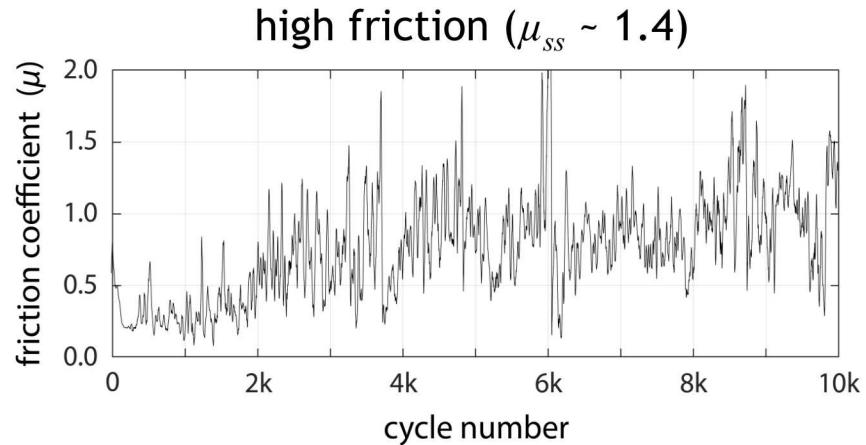
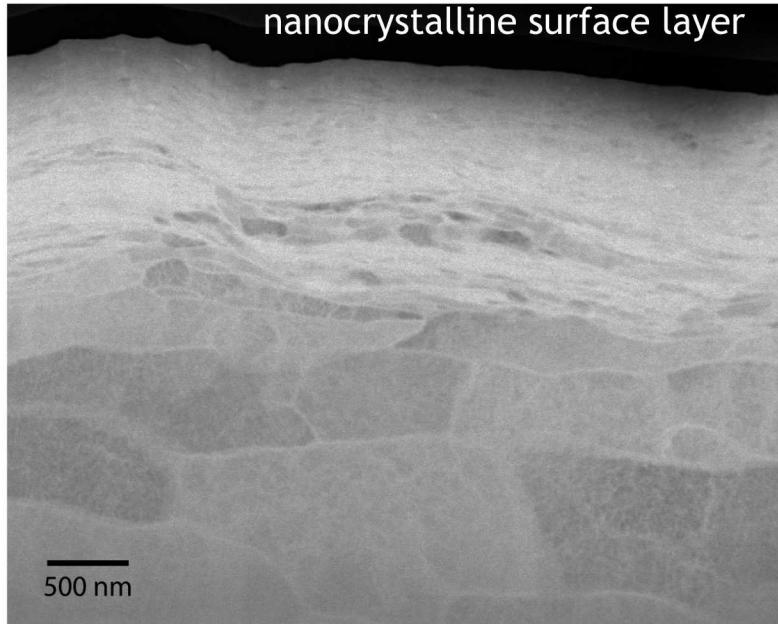
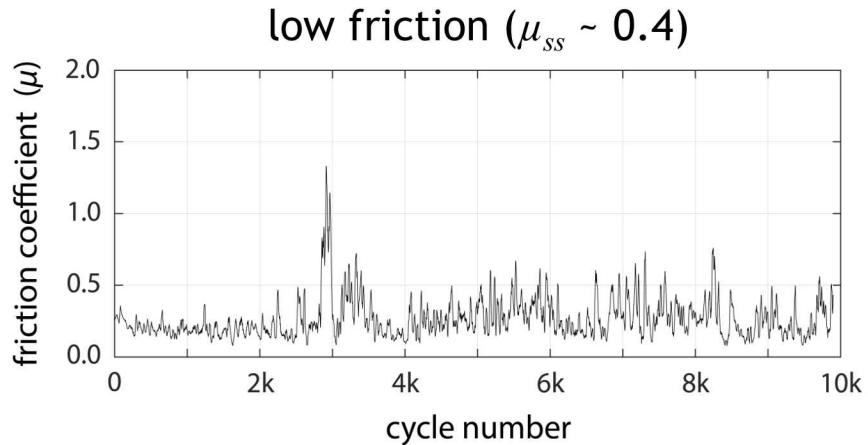
Low friction of FCC metals (pure Cu) in ultra-high vacuum



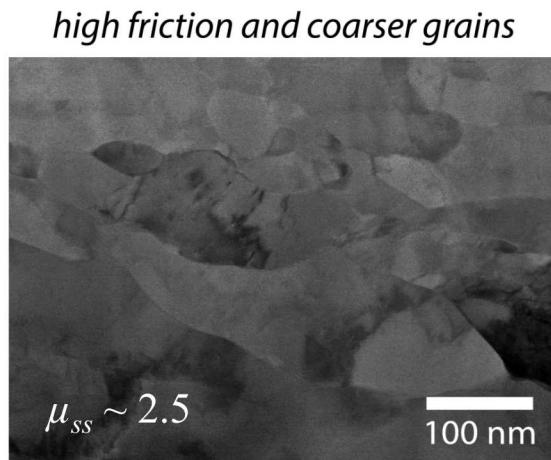
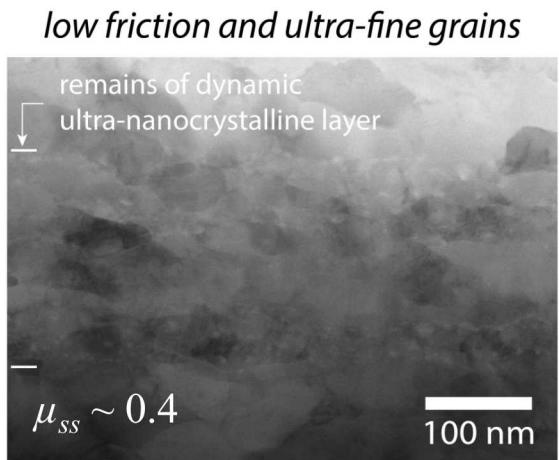
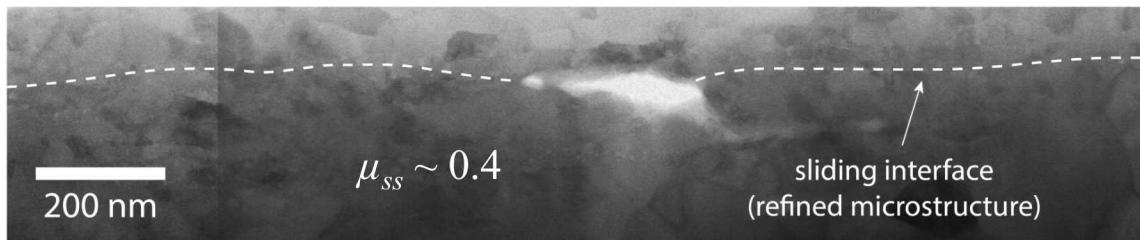
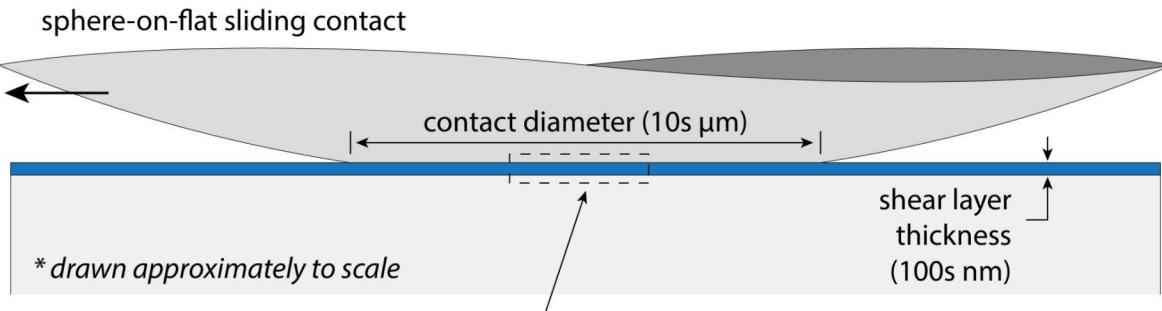
refinement



Low friction of BCC metals (pure Ta) in ultra-high vacuum

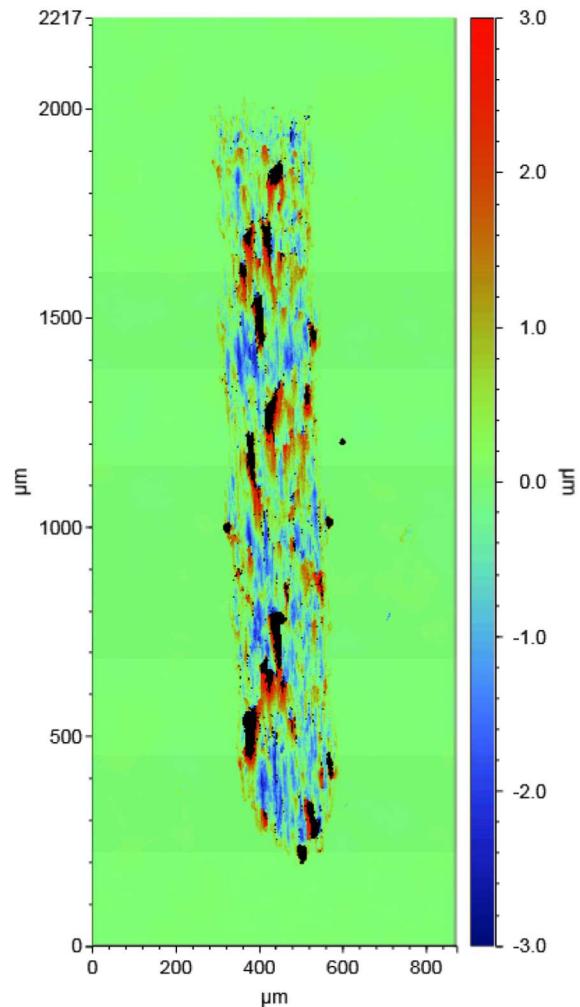


A cross-sectional view of metal sliding contacts



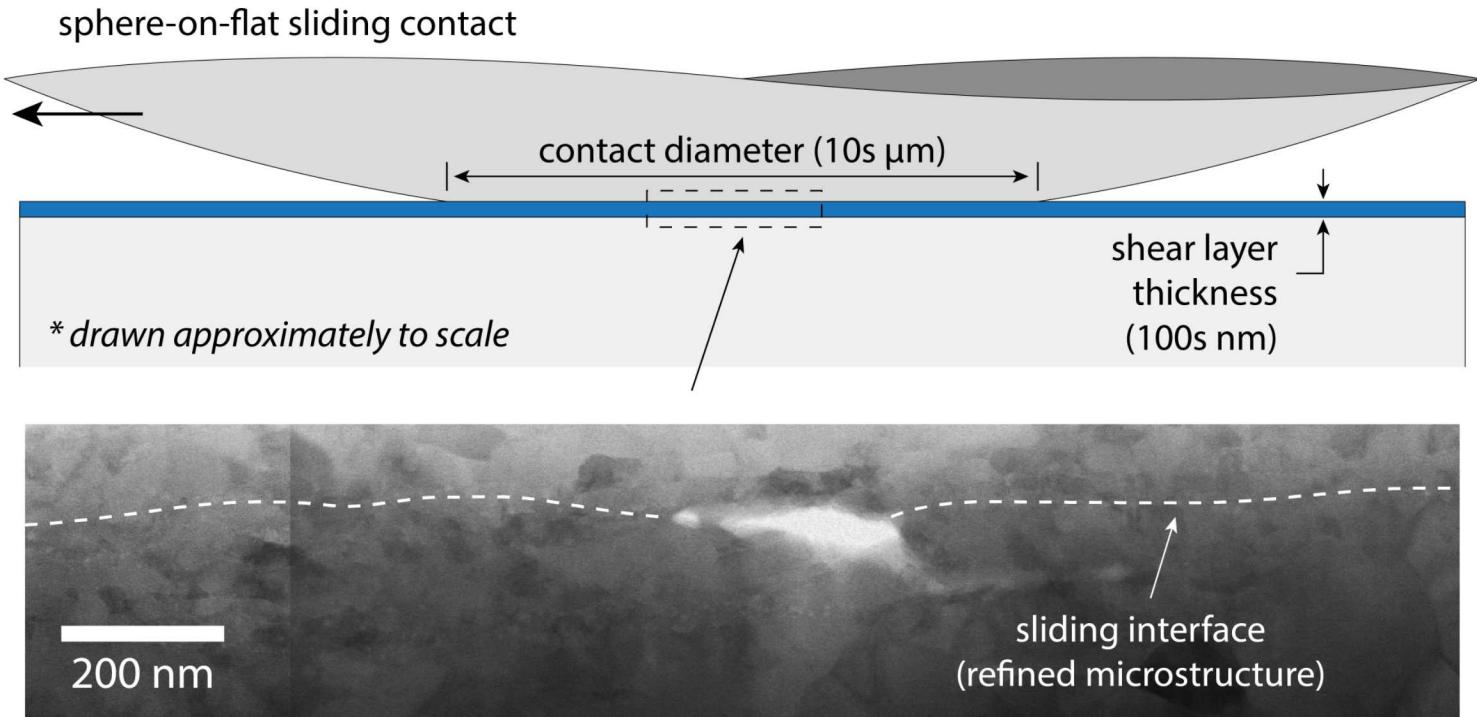
(TEM of self-mated pure Cu in dry N_2)

wear track topography map:



Contact diameters ~ 10s μm
Shear layers ~ 10s nm thick

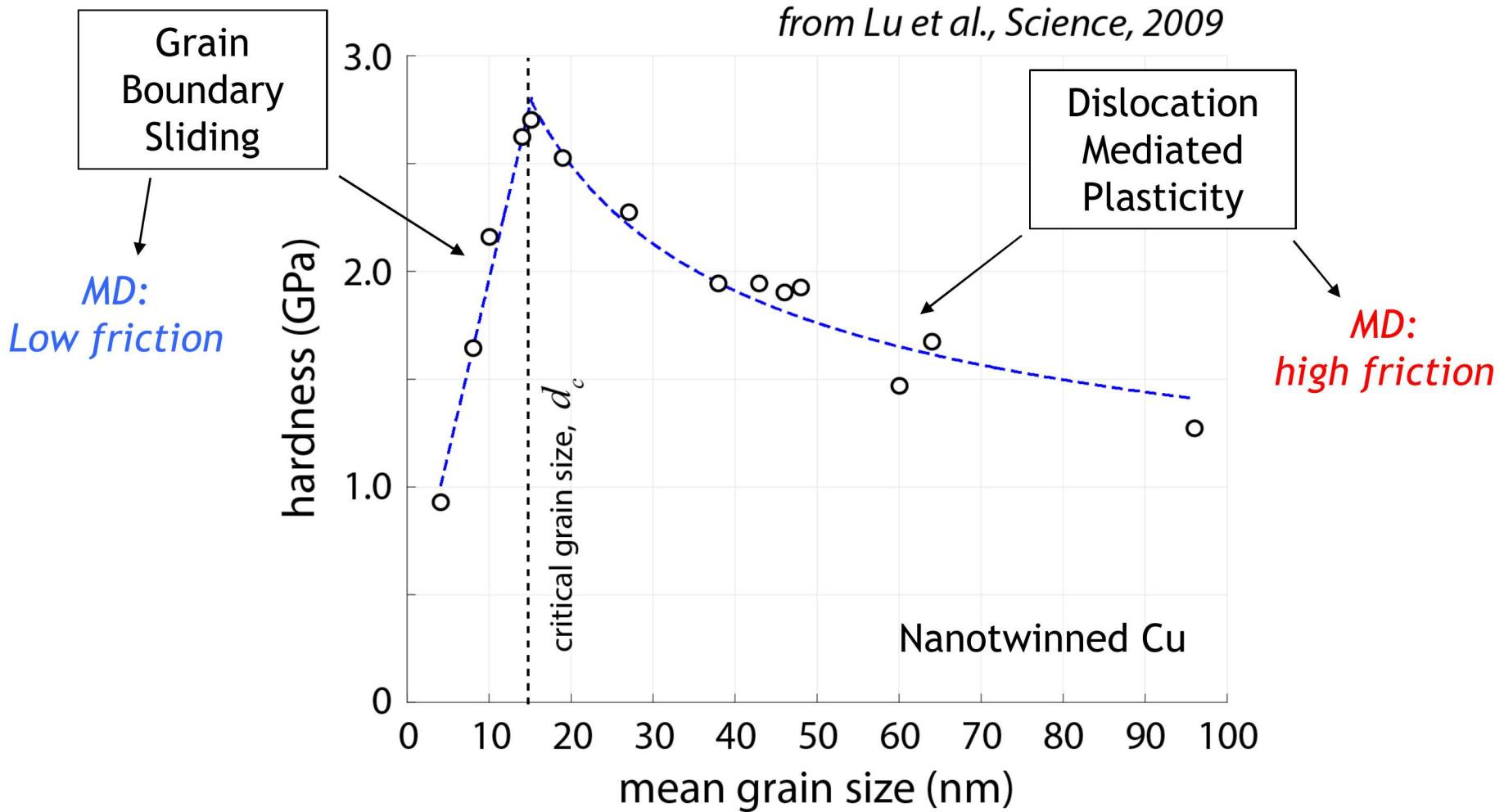
The basis of friction in metals



For metals, the friction coefficient can be described as the ratio of interface shear strength and bulk hardness:

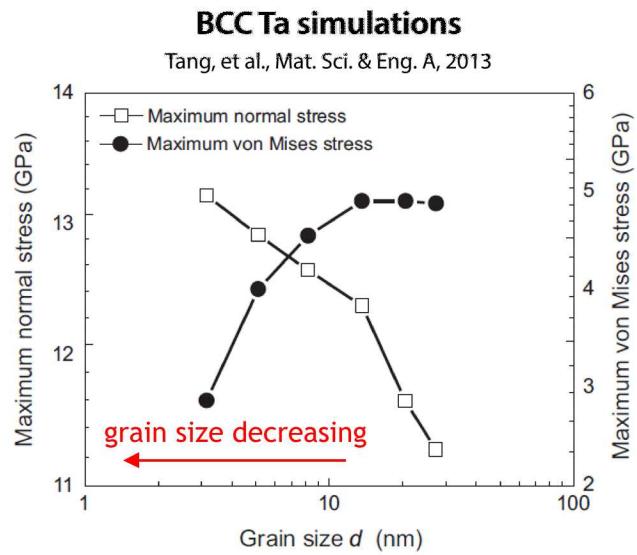
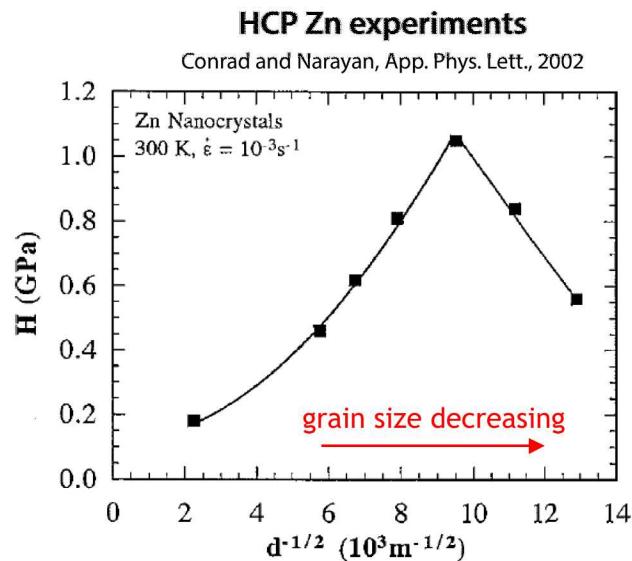
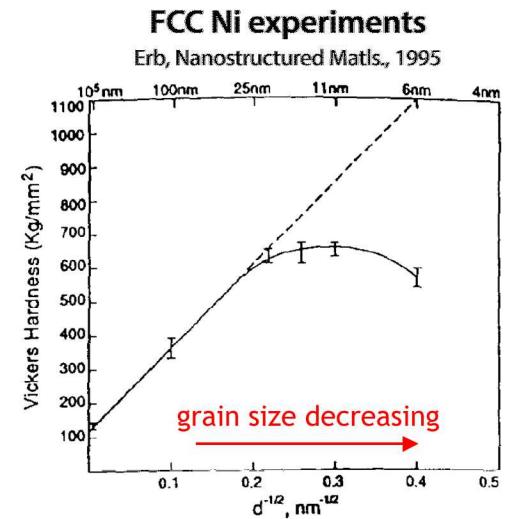
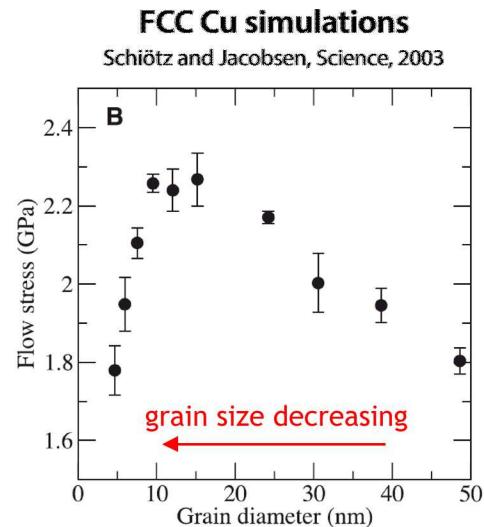
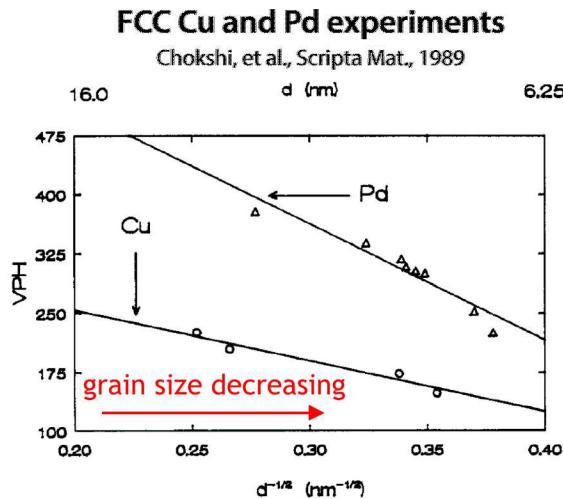
$$\mu = \frac{\tau}{H}$$

How does grain size affect materials properties?



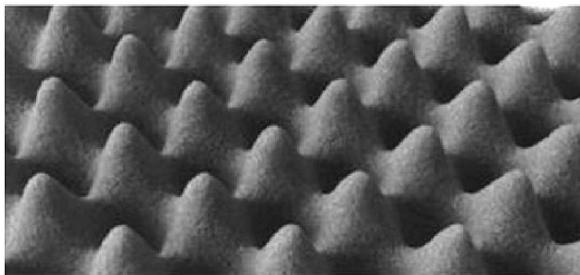
Details of MD simulations can be found in Argibay, et al., *J. Mat. Sci.* (2017)

Hall-Petch breakdown occurs at about 10 nm (critical grain size)

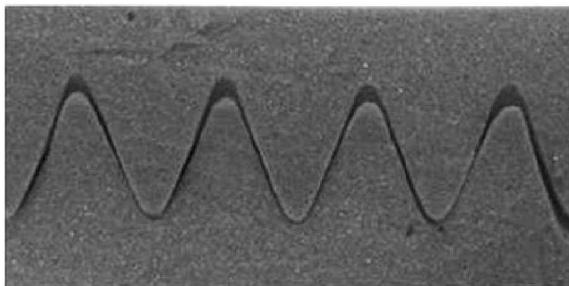


The Mechanism Behind Low and High Friction in Metals

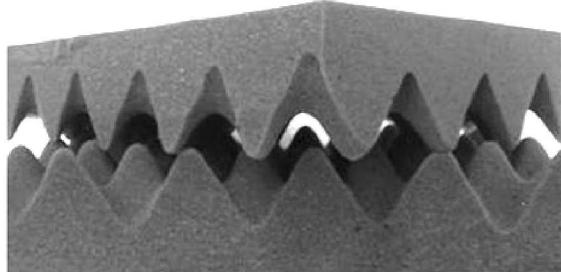
Atomic lattices have periodicity



Commensurate interface
(higher energy barrier/higher μ)



Incommensurate interface
(lower energy barrier/low μ)



High resolution TEM

grain boundary
(incommensurate interface)



twin boundary/atomic lattice
(commensurate interface)

5 nm

Evidence of Intragranular Amorphization

From Kaibyshev, *Mat. Sci. & Eng.*, 2002:

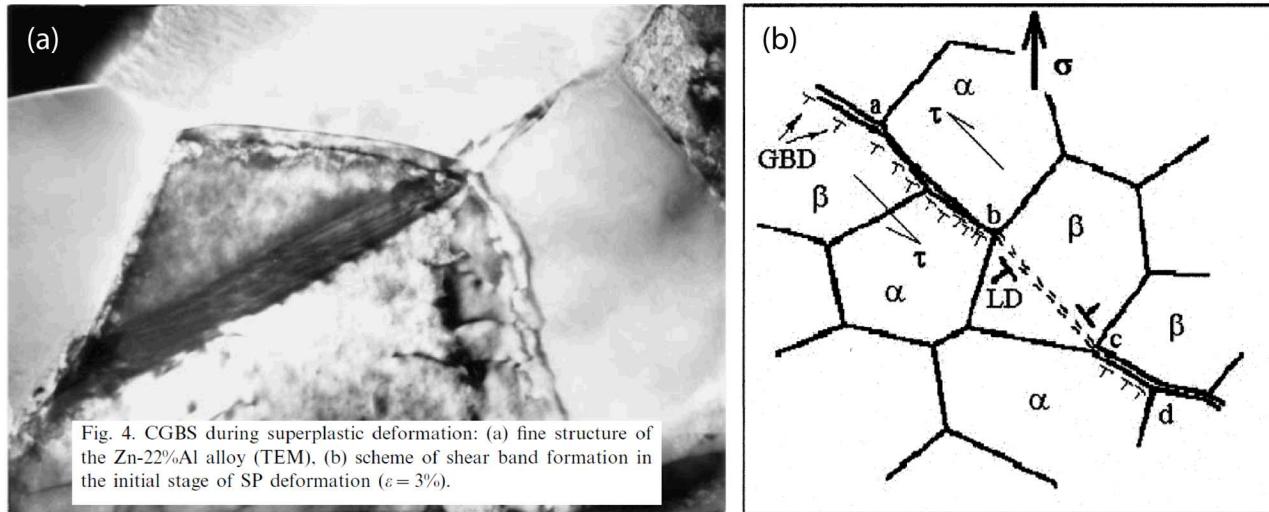
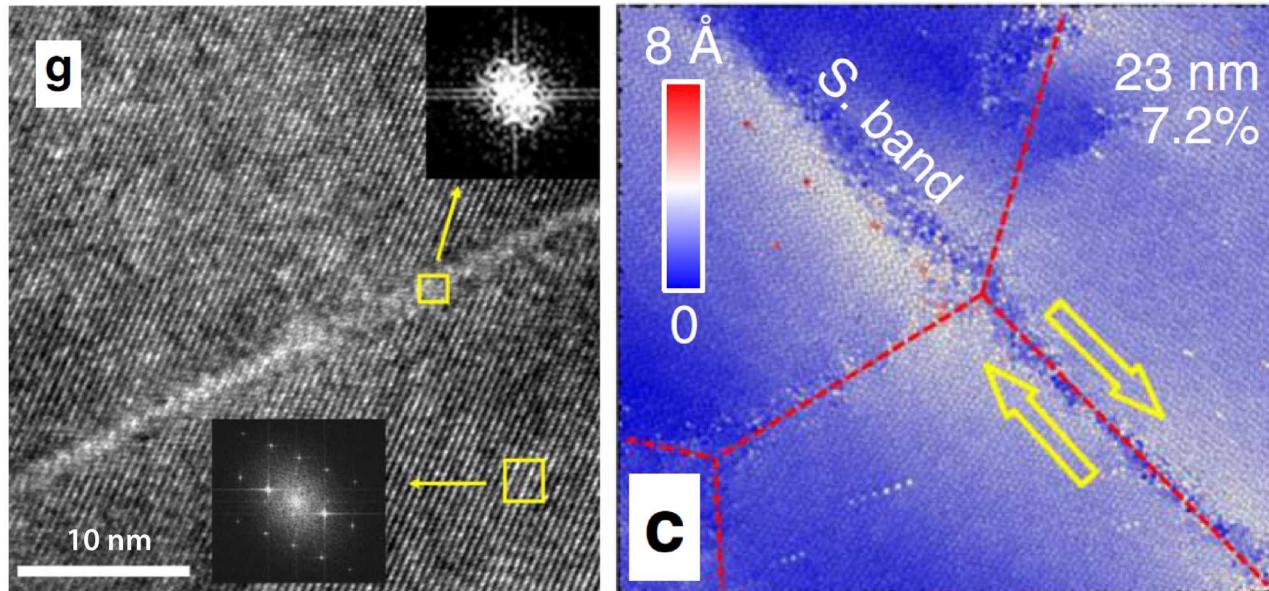
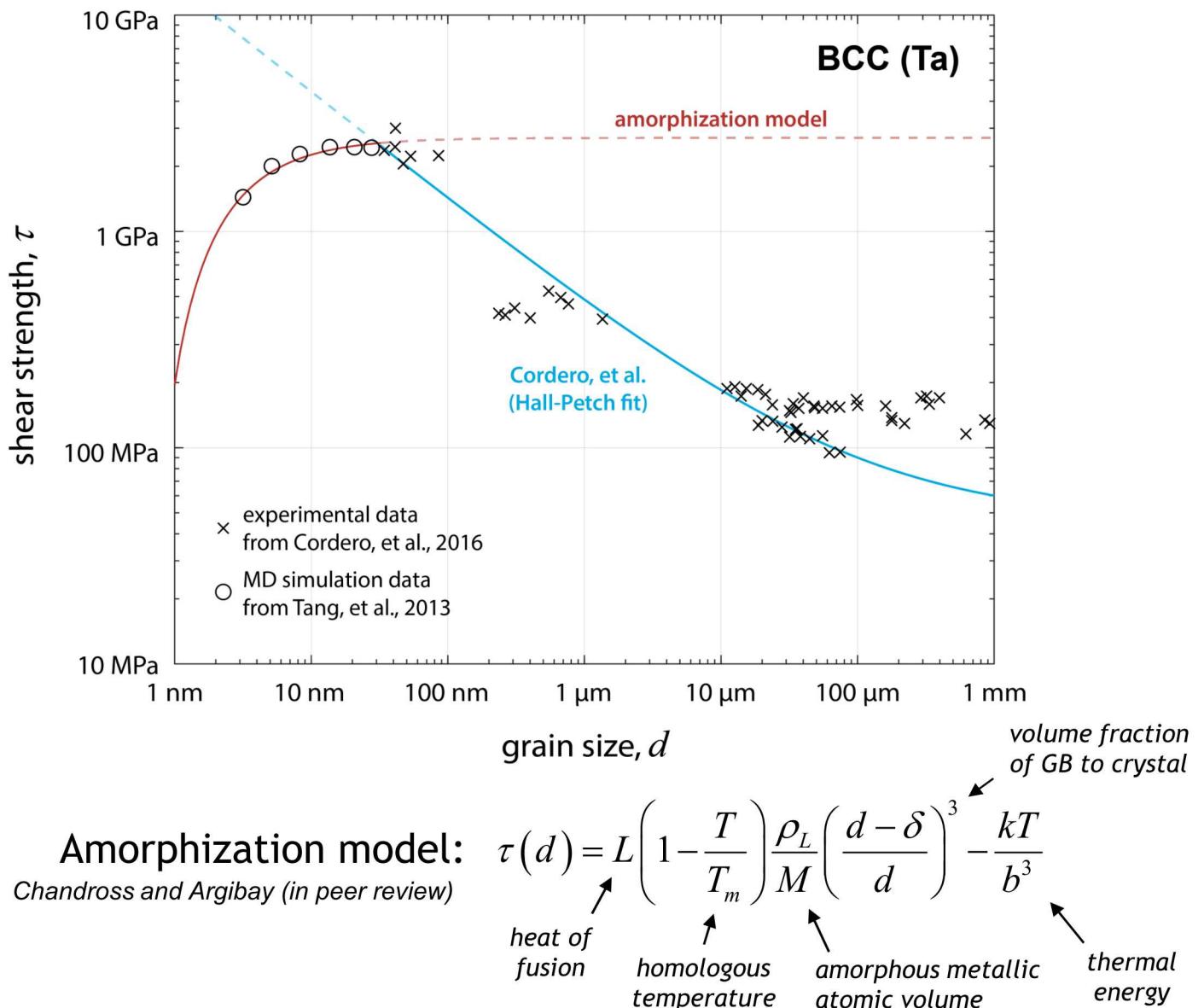


Fig. 4. CGBS during superplastic deformation: (a) fine structure of the Zn-22%Al alloy (TEM), (b) scheme of shear band formation in the initial stage of SP deformation ($\varepsilon = 3\%$).

From Luo, et al., *Nat. Comm.*, 2019:



Grain Size Dependent Strength of Metals



Rosenhain and Ewen predict amorphous

THE INTERCRYSTALLINE

MATERIALS

(SEC

By WALTER ROSENHAIN,

DONAL

(BOTH OF THE NAT

IN their first paper on the intercrystalline layer, Rosenhain and Ewen put forward what they believed to be the first evidence in favour of an hypothesis that the cohesion in metals is due to the presence of an intercrystalline layer which may be briefly termed the "amorphous or non-crystalline metal". They suggested that the crystals of which the metal is composed are "cemented" together by a layer of amorphous or non-crystalline metal. The substance of the metal or the intercrystalline layer is regarded as being at least as closely analogous to undercooled liquid as to the minute interstices which meet one another in various

Slip at Grain Boundaries and Grain Growth in Metals

By N. F. MOTT,
H. H. Wills Physical Laboratory, Bristol

(b) The observed fact that at the melting point the slip is the same as that which would be given by a monomolecular layer of liquid appears in the theory as an *accident*. The mechanism of flow in liquid aluminium cannot be anything like that sketched here, because the temperature dependence is 10–20 times smaller; the viscosity of liquid metals depends on temperature according to the formula

$$\sigma = \sigma_0 e^{W/kT},$$

where W is of the order of the latent heat of fusion (Frenkel 1946).

We must therefore modify our hypothesis. Let us suppose that the elementary act which allows slip to occur is the *disordering* of atoms round each island where it is good. The free energy F necessary to do this will approach zero at the melting point and nL at the absolute zero of temperature; here L is the latent heat of fusion per atom. At any other temperature, let us assume F to be given by

$$F = nL(1 - T/T_M),$$

where T_M is the temperature of melting. Let us assume also, since the disordering will result in a slip through a distance a , that a stress σ will decrease or increase F by $\pm \frac{1}{2} \sigma n \omega a$. Then the rate of slip is now

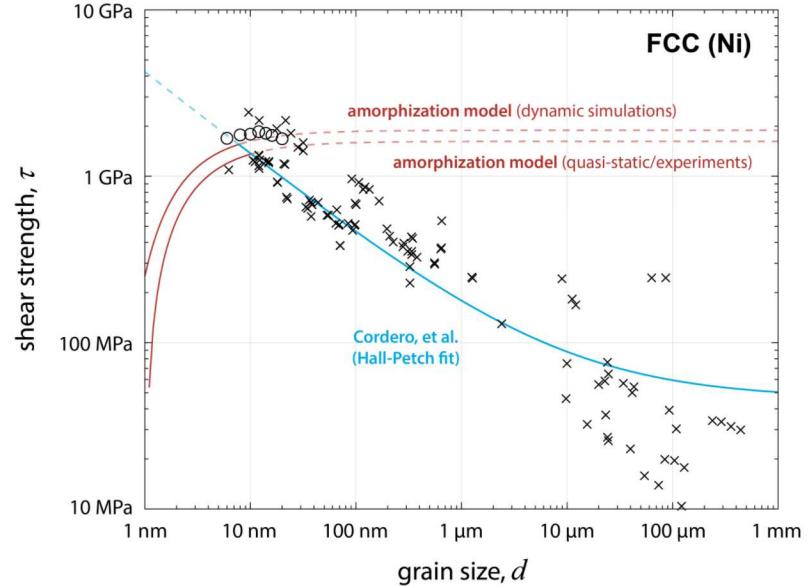
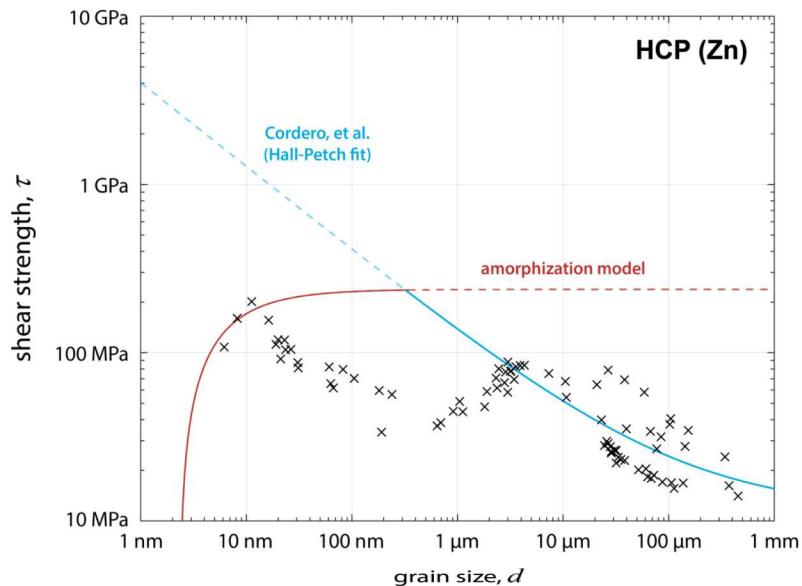
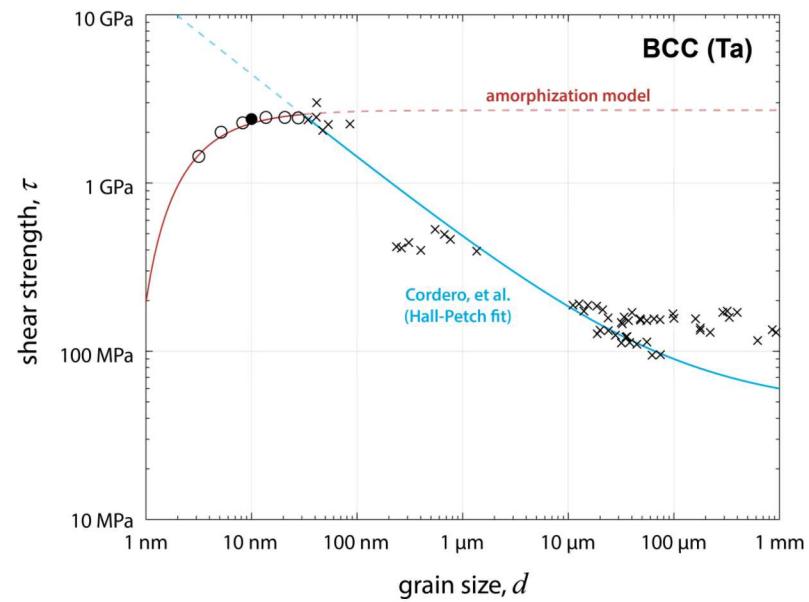
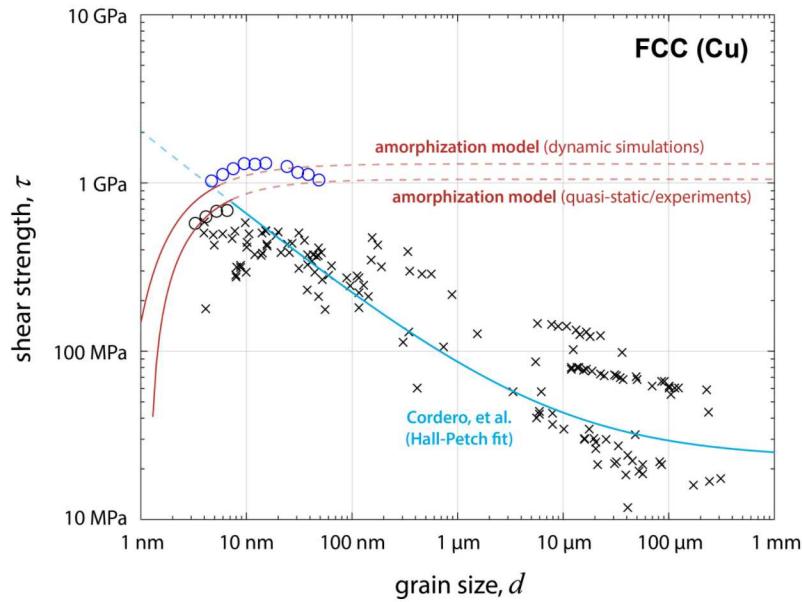
$$v = 2\pi a \exp \{ -nL(1 - T/T_M)/kT \} \sinh(\sigma n \omega a / 2kT),$$

which for small σ reduces to

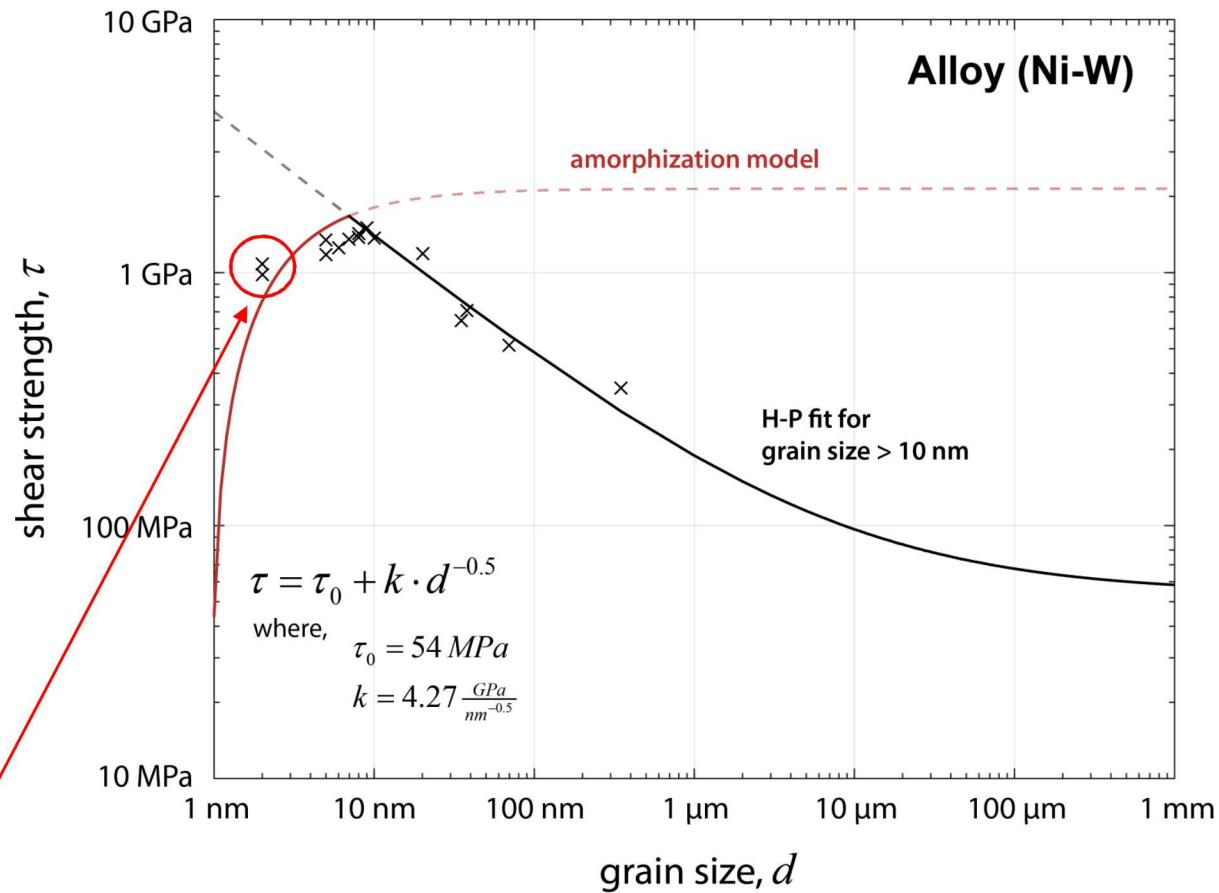
$$v = \frac{\pi a^2 n \omega \sigma}{kT} \exp \left(\frac{nL}{kT_M} \right) \exp \left(\frac{-nL}{kT} \right).$$

Mott provided a theory of slip based on the formation of disordered, liquid-like islands of atoms

The model works well for many other metals



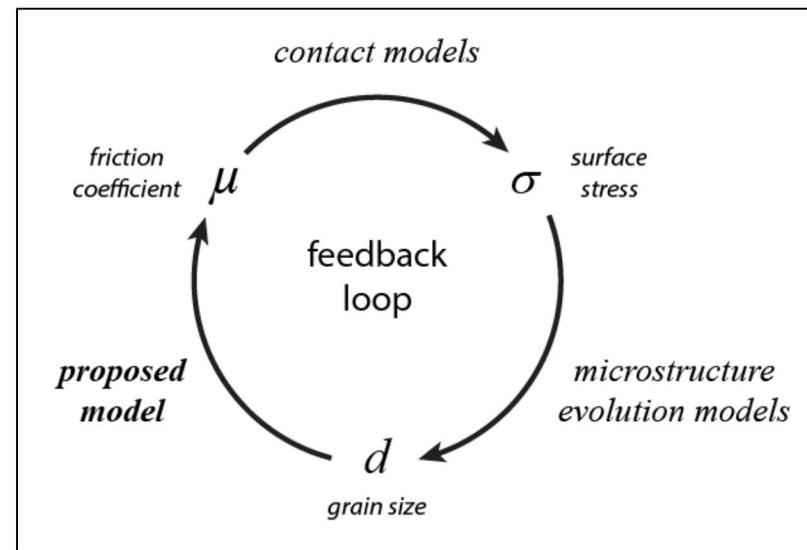
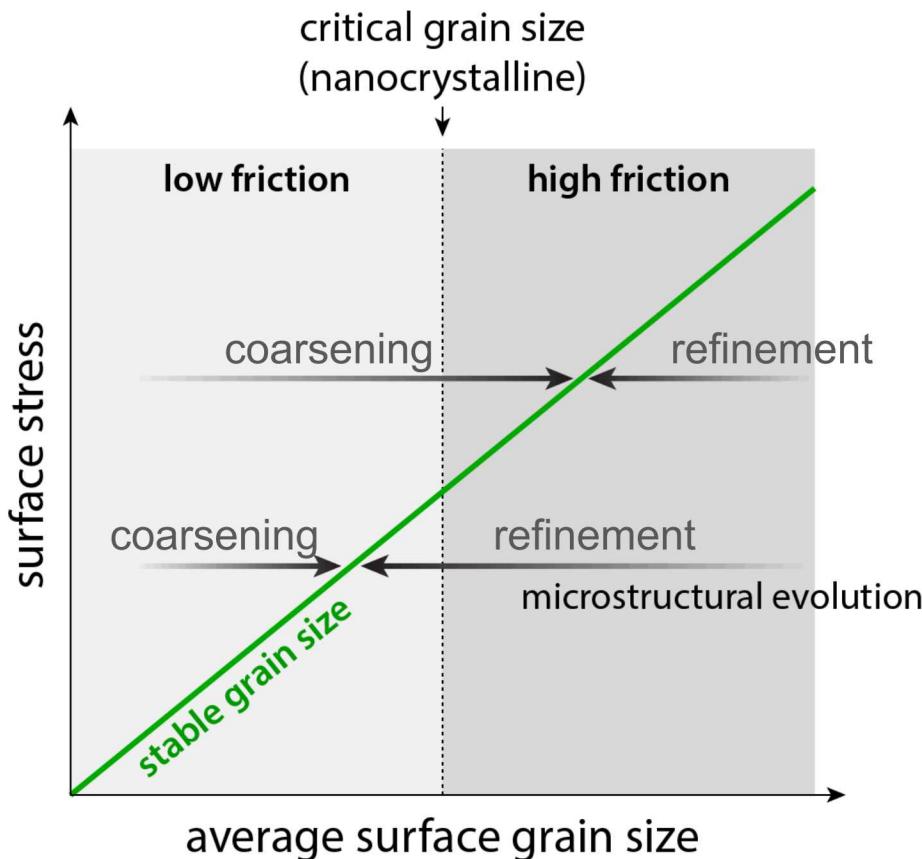
Amorphization model also works for alloys



Keep in mind:

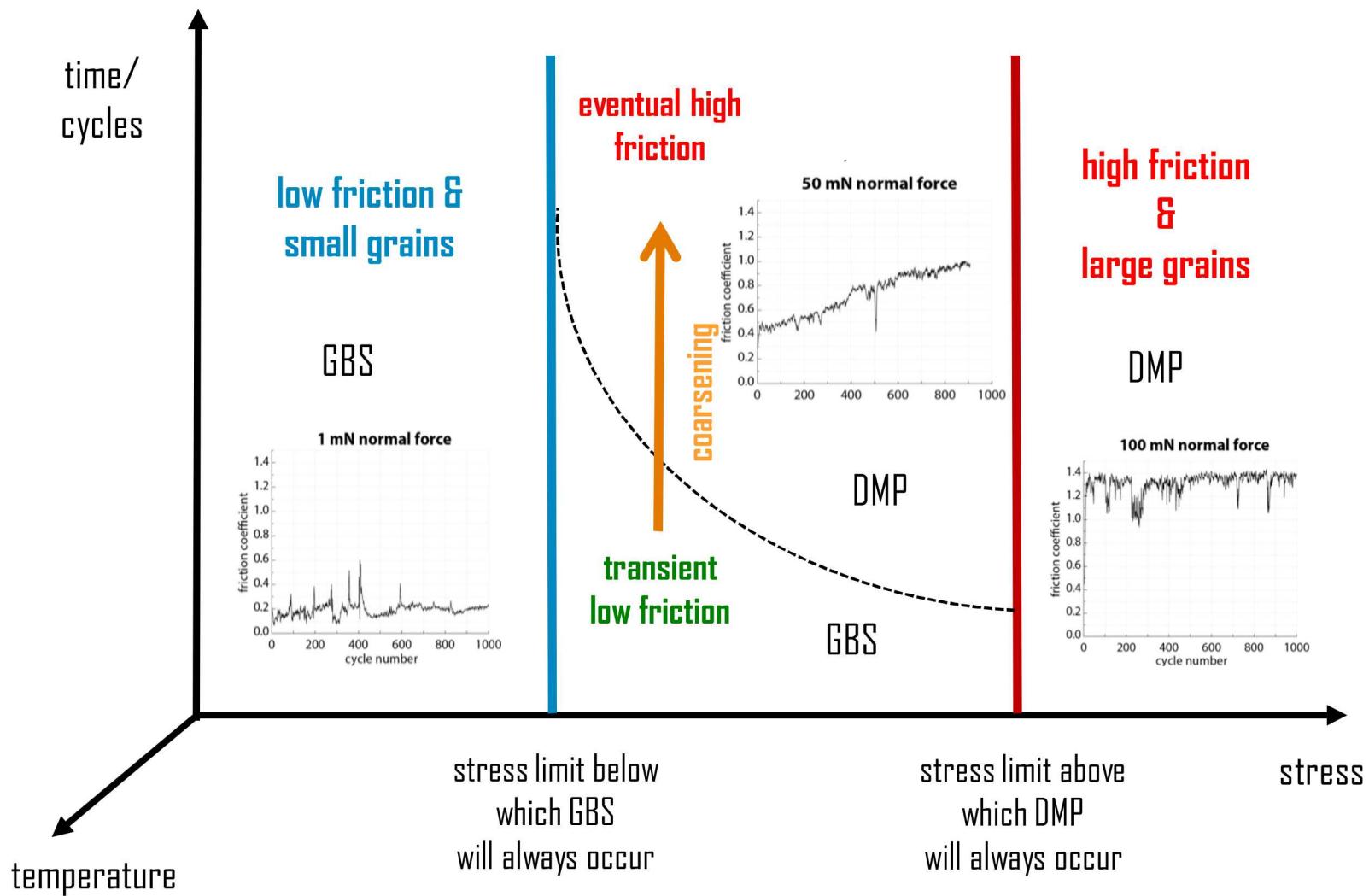
1. Grain size estimates are particularly difficult and exhibit higher uncertainty in the 1-10 nm range, and
2. these errors are relatively small compared to typical H-P fit for far coarser grain size (e.g. earlier pure metal data)

Proposed stress-dependent steady-state (asymptotic) grain size

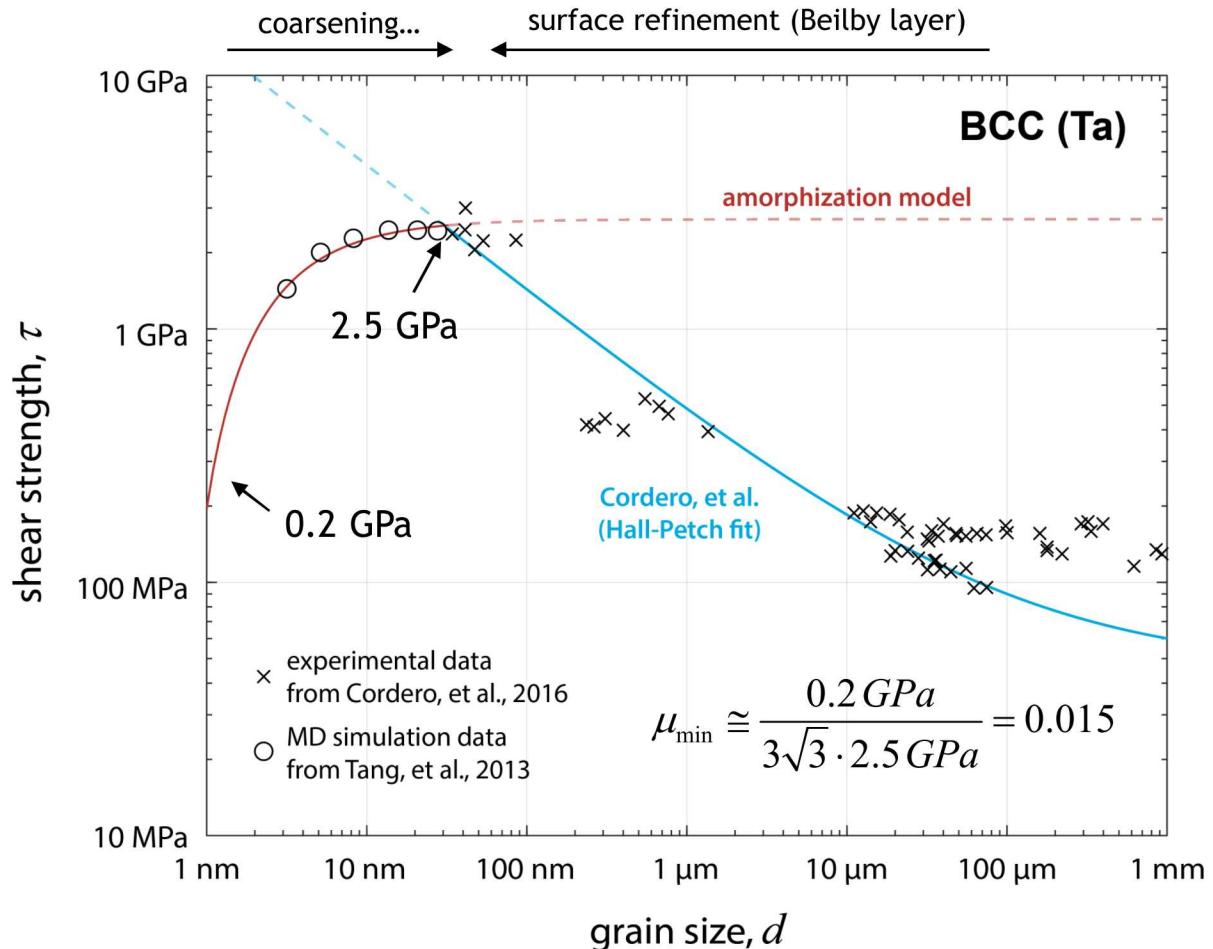


- Effective refinement from recrystallization (Zener & Holloman, 1944; McQueen et al., 1967)
- Known in rocks and ice cores (Derby et al., 1992)
- Recently extended to metals under severe plastic deformation (Pougis et al., 2014)

Generalized friction regimes map for metals

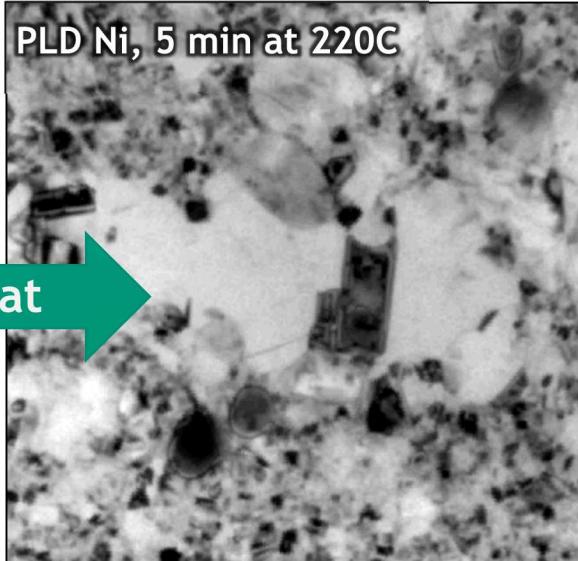
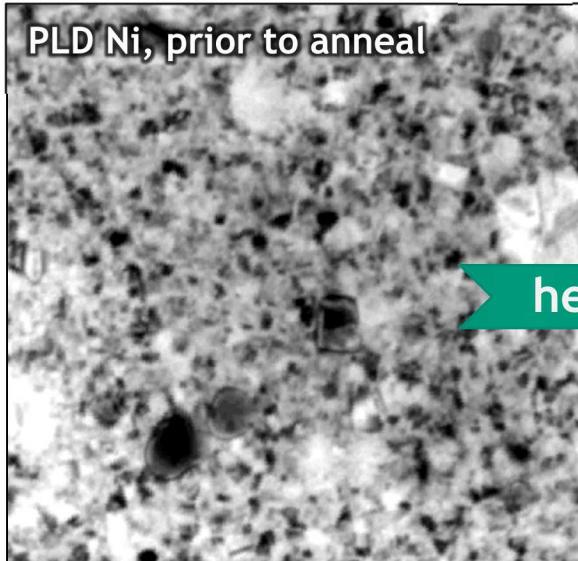


What is the lowest friction condition?



Bowden & Tabor: $\mu = \frac{\tau}{H} \xrightarrow{\text{Tabor}} \mu = \frac{\tau}{3 \cdot \sigma_Y} \xrightarrow{\text{Von Mises}} \mu = \frac{\tau(d_{\text{surface}})}{3\sqrt{3} \cdot \tau(d_{\text{bulk}})}$

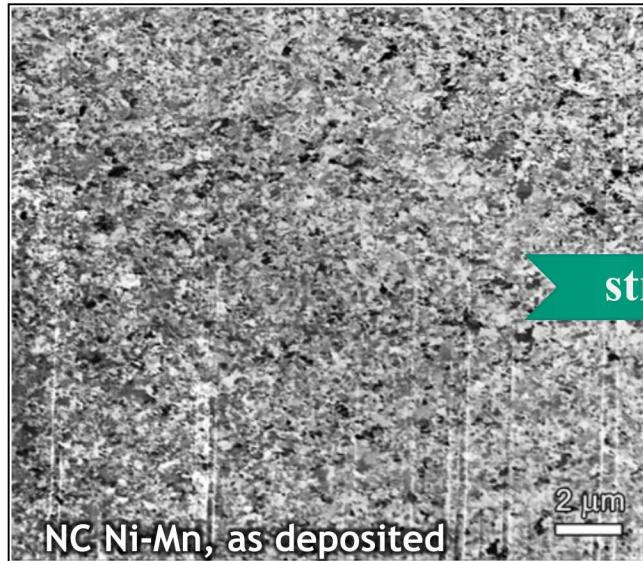
The challenge of grain size instability...



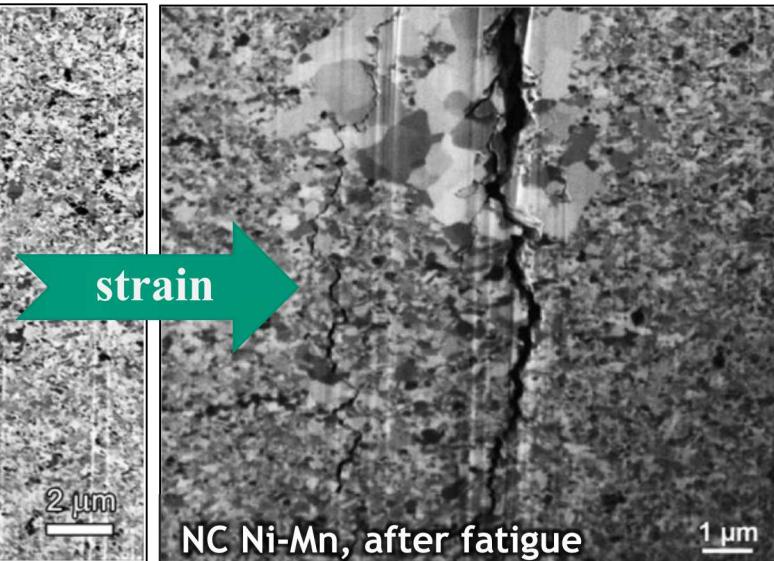
heat

Grain growth in nanocrystalline metals is driven

Thermally...



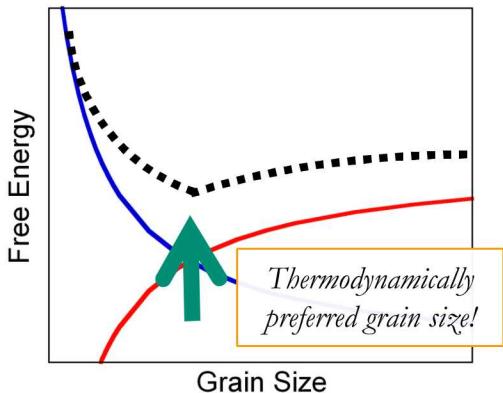
...and Mechanically



strain

Regular Nanocrystalline Solution Model (Chris Schuh Group at MIT)

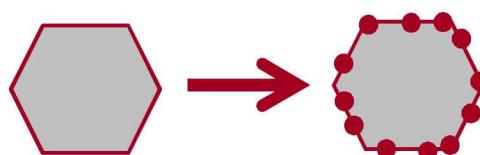
Slide courtesy of C. Schuh and H. Murdach (MIT)



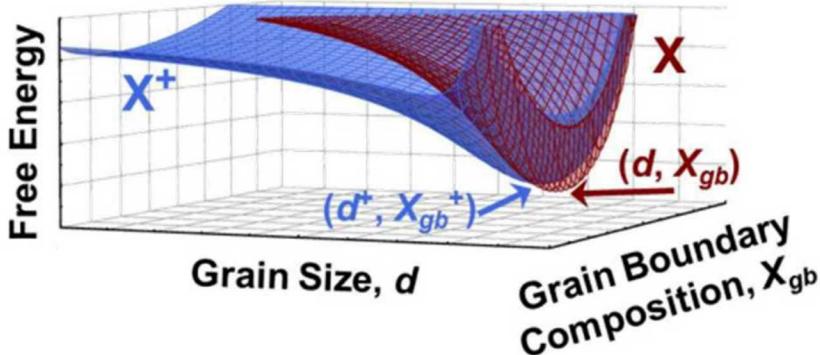
RNS Model:

Binary metal alloys exist possessing highly (intrinsically?) thermodynamically stable nanocrystallinity.

Solute segregation at grain boundaries:



$$dG = \left[\gamma - \frac{N_\beta}{A} \Delta G_{seg} \right] dA$$



References:

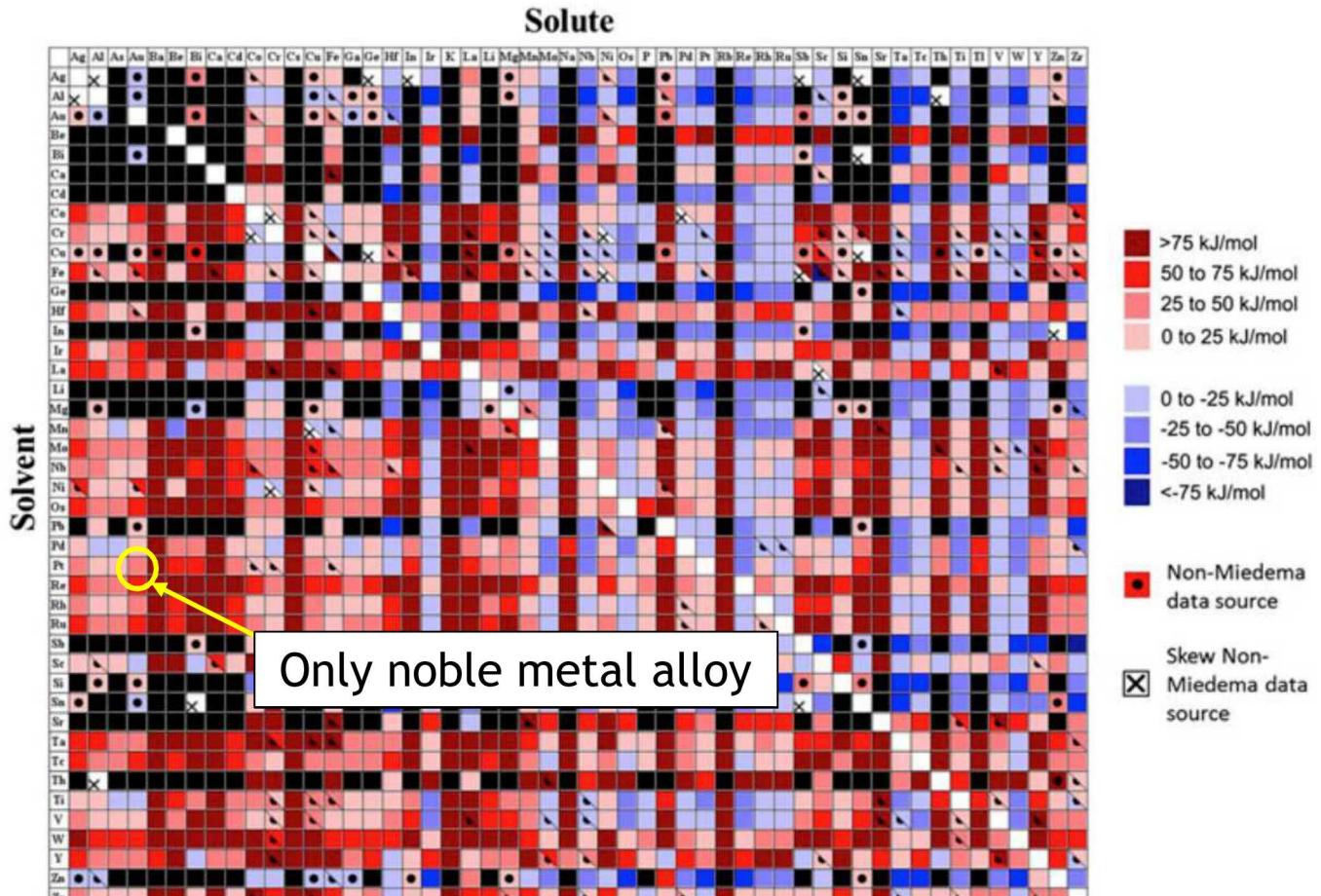
T. Chookajorn, et al., *Science*, 2012

Kirchheim, *Acta Materialia*, 2002

Weissmuller, *J. Materials Research*, 1994

D.S. Gianola et al., *Acta Materialia*, 2006

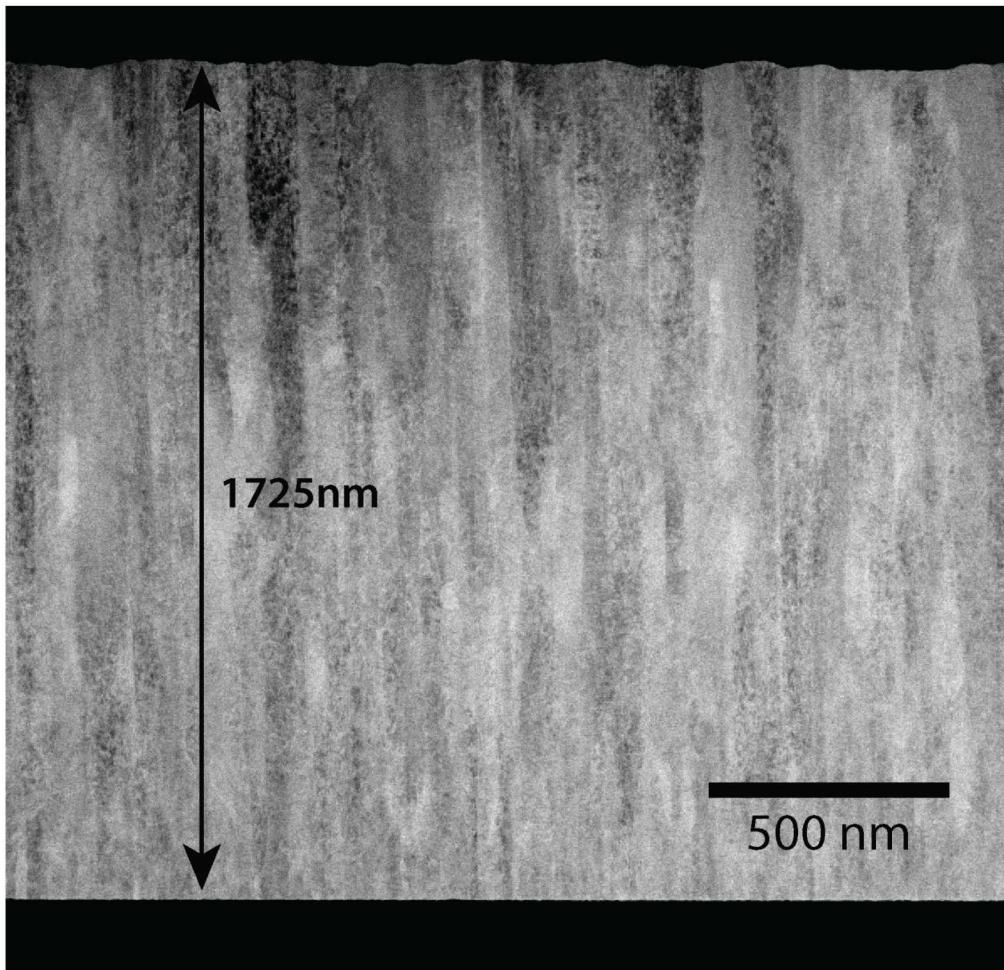
So many alloys to choose from!



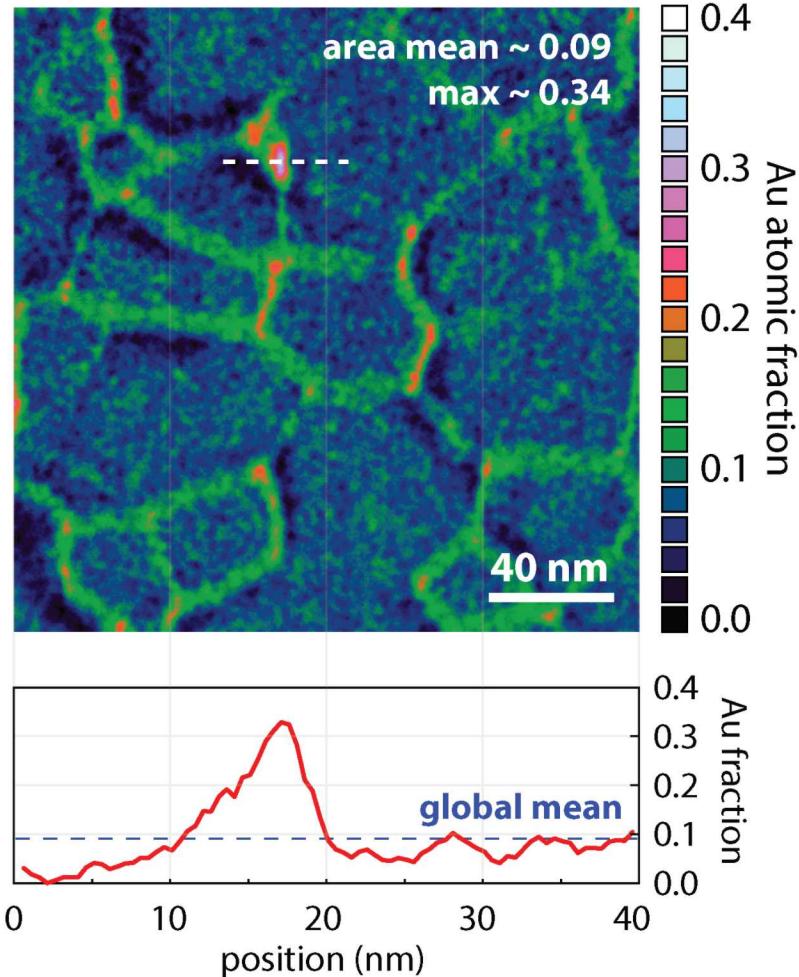
Murdoch & Schuh, *J. Mater. Res.* 2013.

Columnar grains, and Au segregates to Pt GBs

A) PtAu cross-section

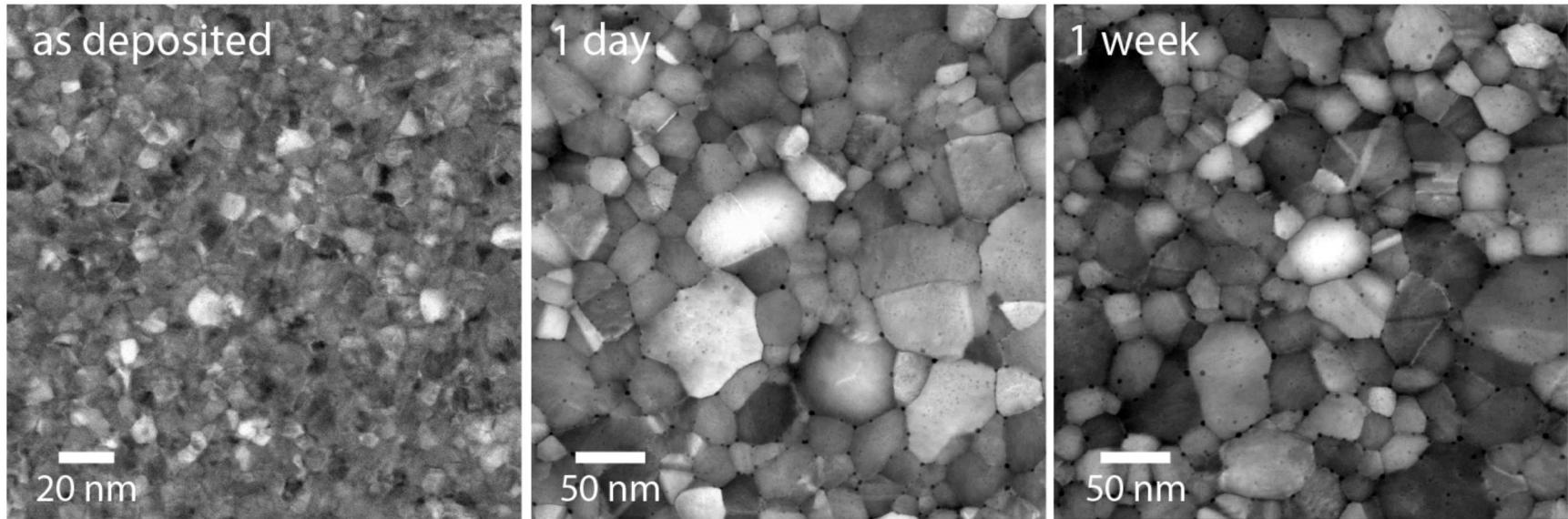


B) Au atomic fraction

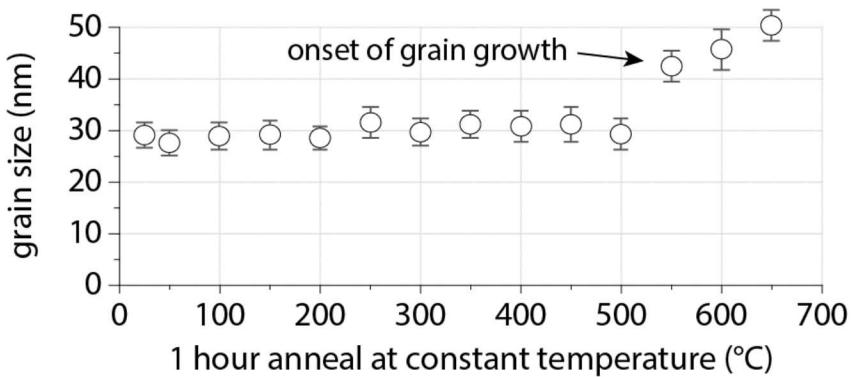


We synthesized coatings of Pt with 10% Au

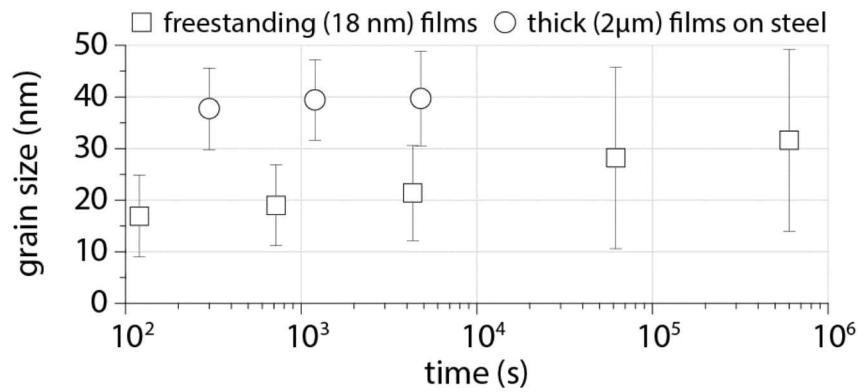
Excellent thermal stability. What about mechanical?



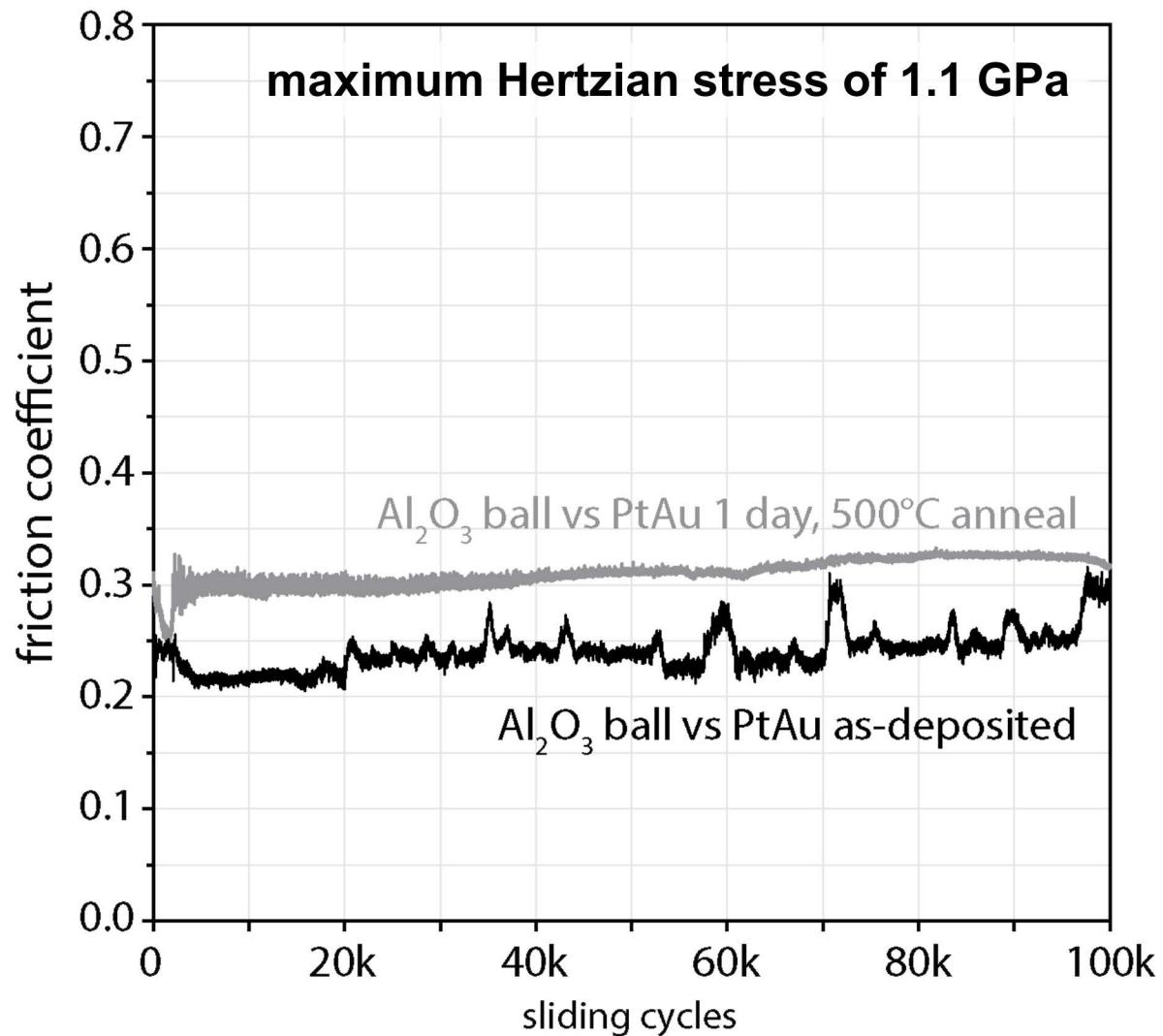
XRD in situ grain growth



TEM in situ grain growth at 500°C

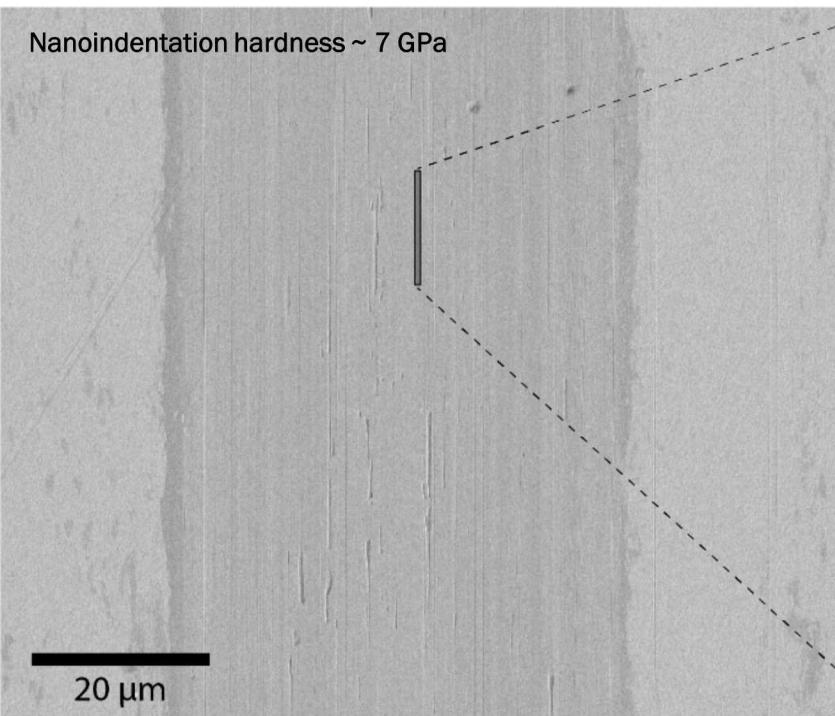


Stable nanocrystalline PtAu exhibited long-lived low friction
(sliding against sapphire)

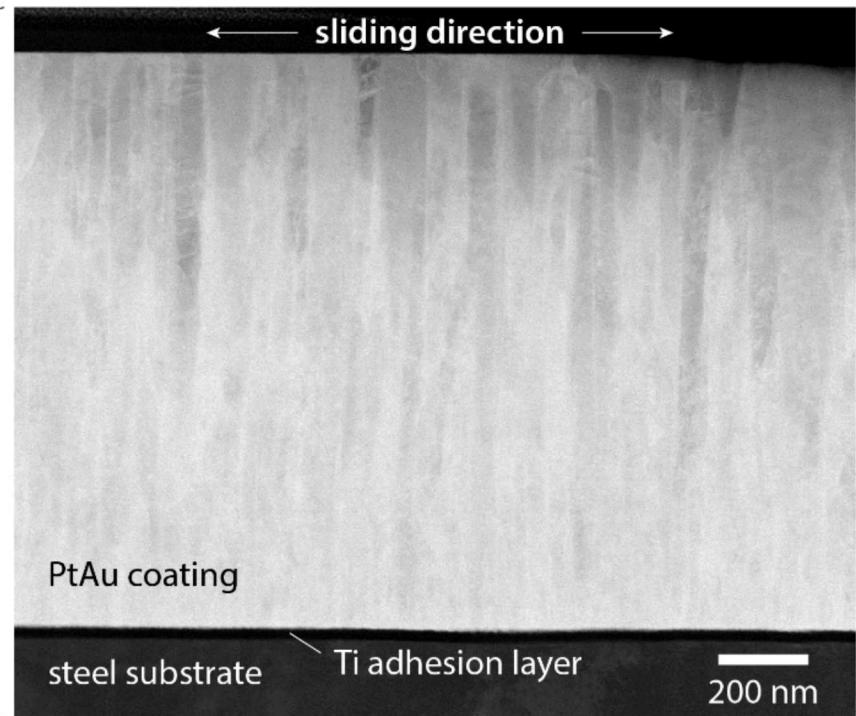


No evidence of microstructural evolution after prolonged tests

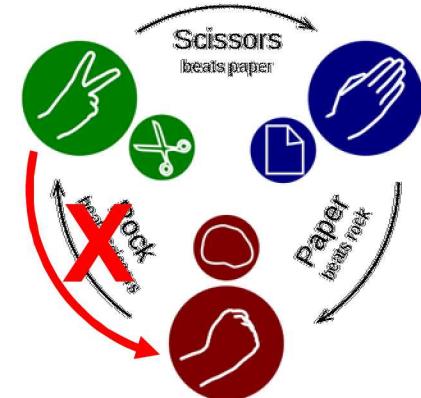
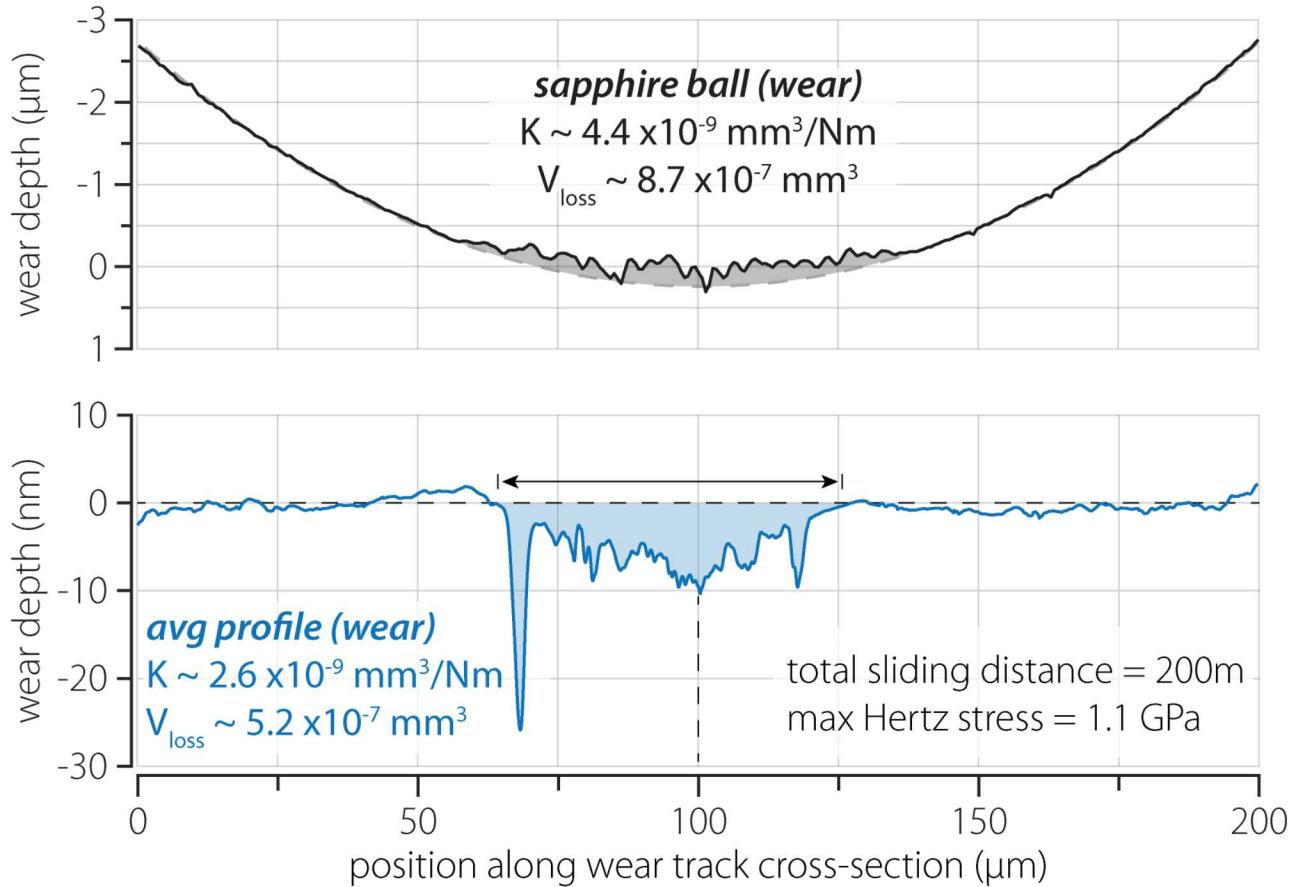
SEM of PtAu wear track after 100k passes



TEM cross-sections of wear track

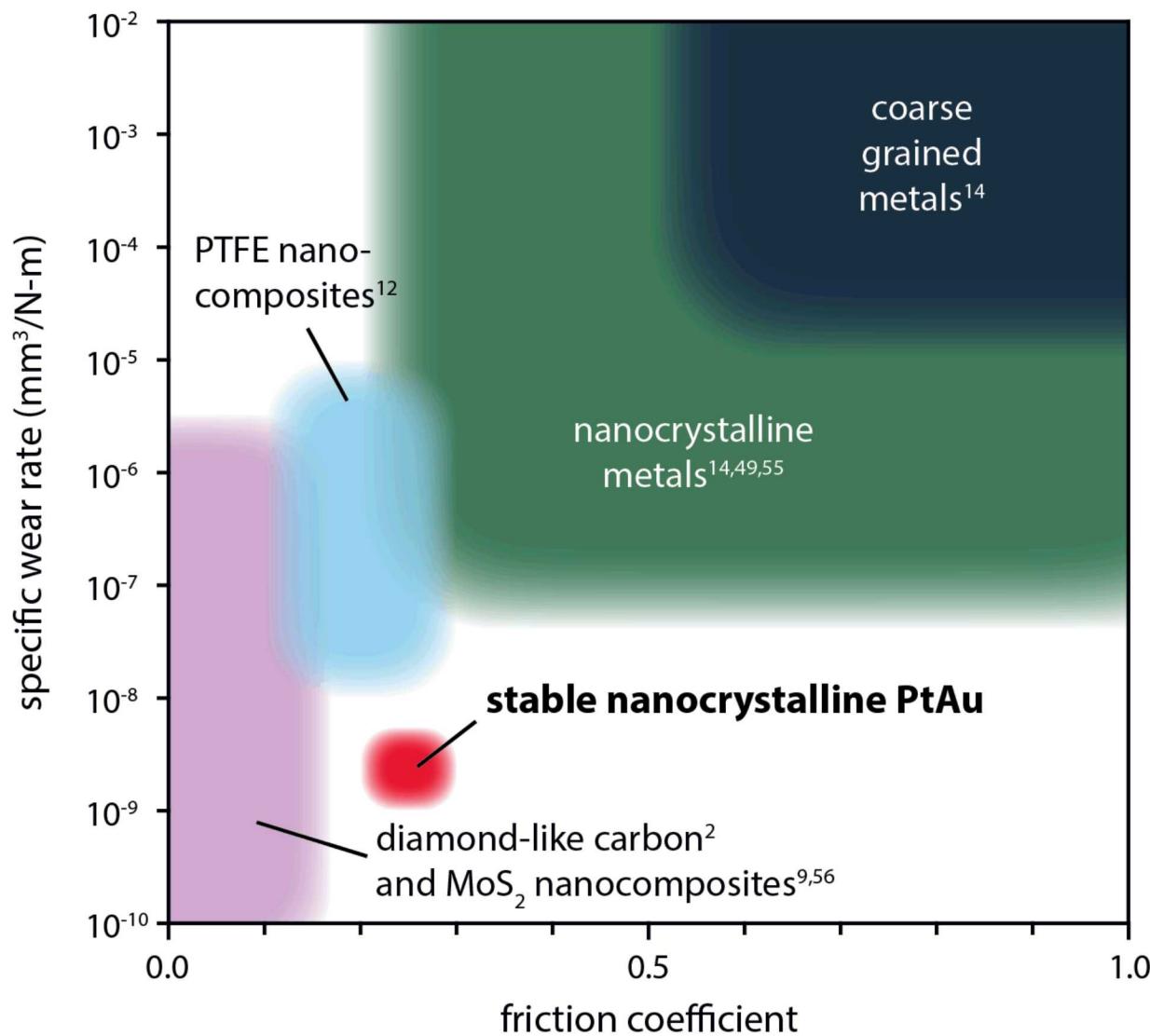


No evidence of microstructural evolution
(maximum Hertzian stress of 1.1 GPa)

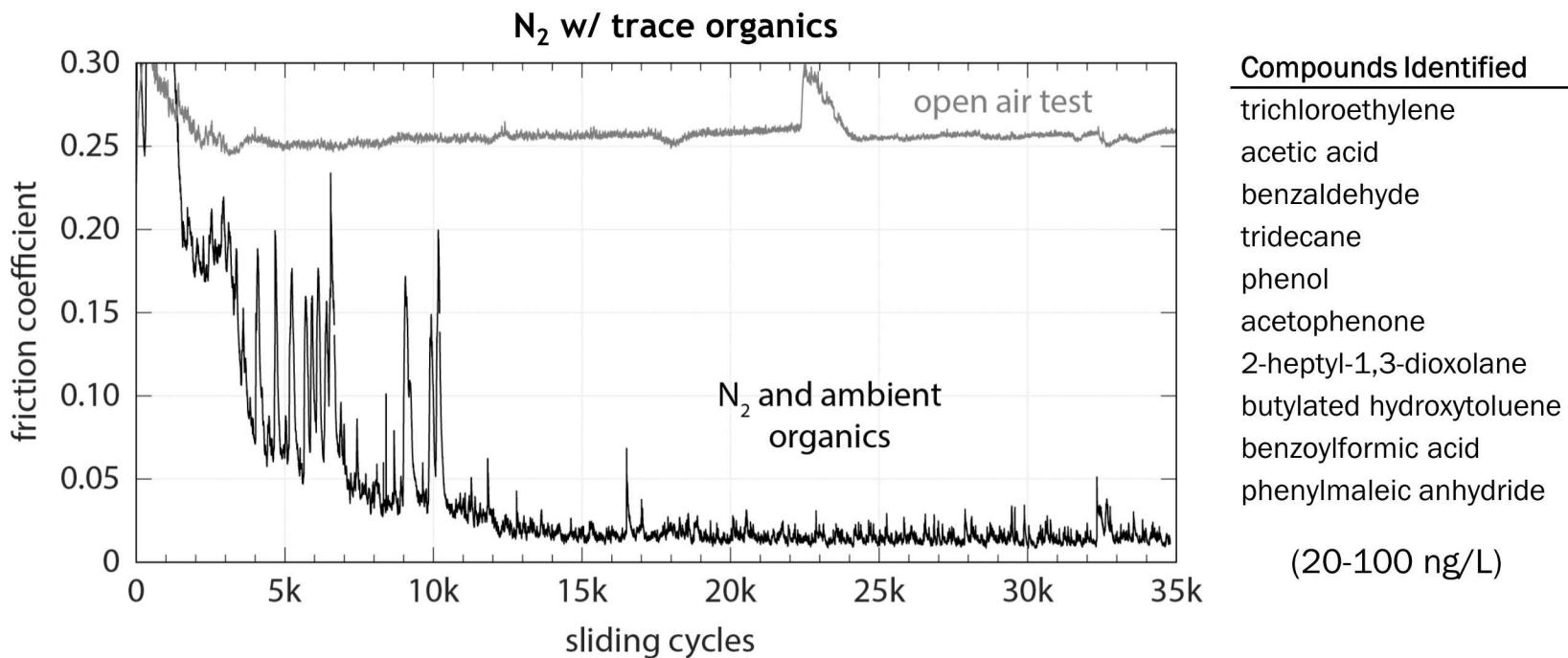
Wear in Cross-Section: Ultra-low wear rate of PtAu (3×10^{-9} mm³/N·m)

... when scissors beat rock!

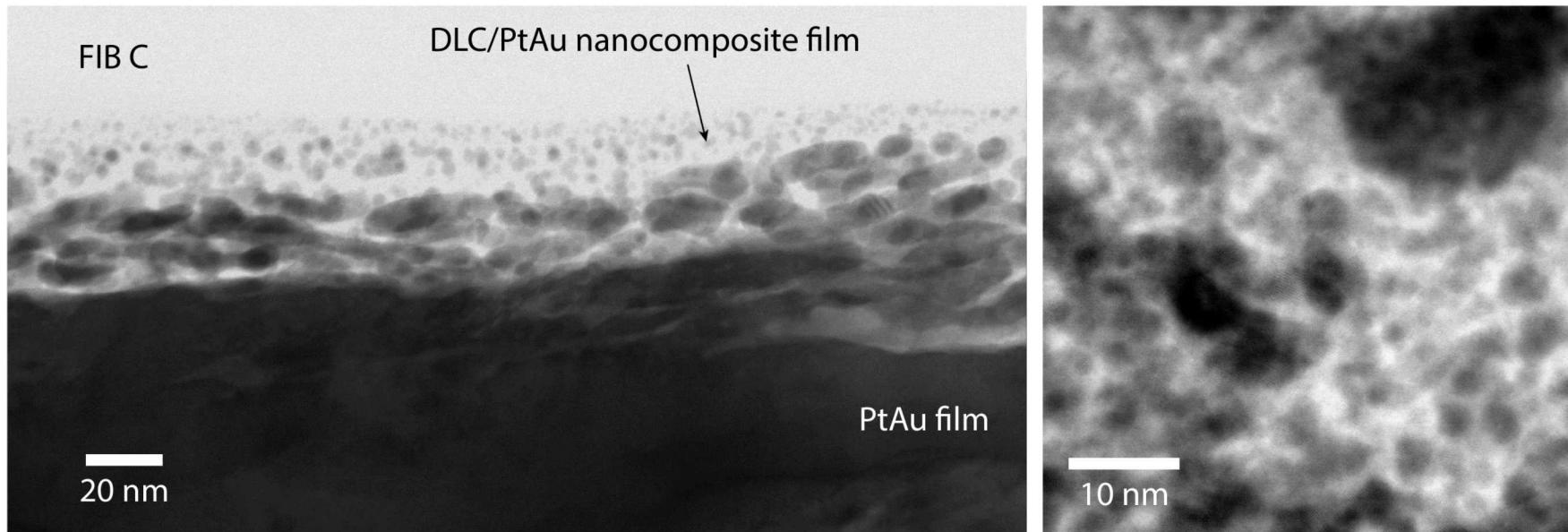
Significant improvement, more work to do...



Friction Behavior in Anaerobic Environments



*... in situ tribo-chemical formation of DLC films from “thin air”!
(Albuquerque is at 6,000ft/1,800m of elevation, after all...)*



confirmed **diamond-like carbon (DLC)** using Raman analysis
with 20% hydrogenation using elastic recoil

Acknowledging the Many Contributors to this Story

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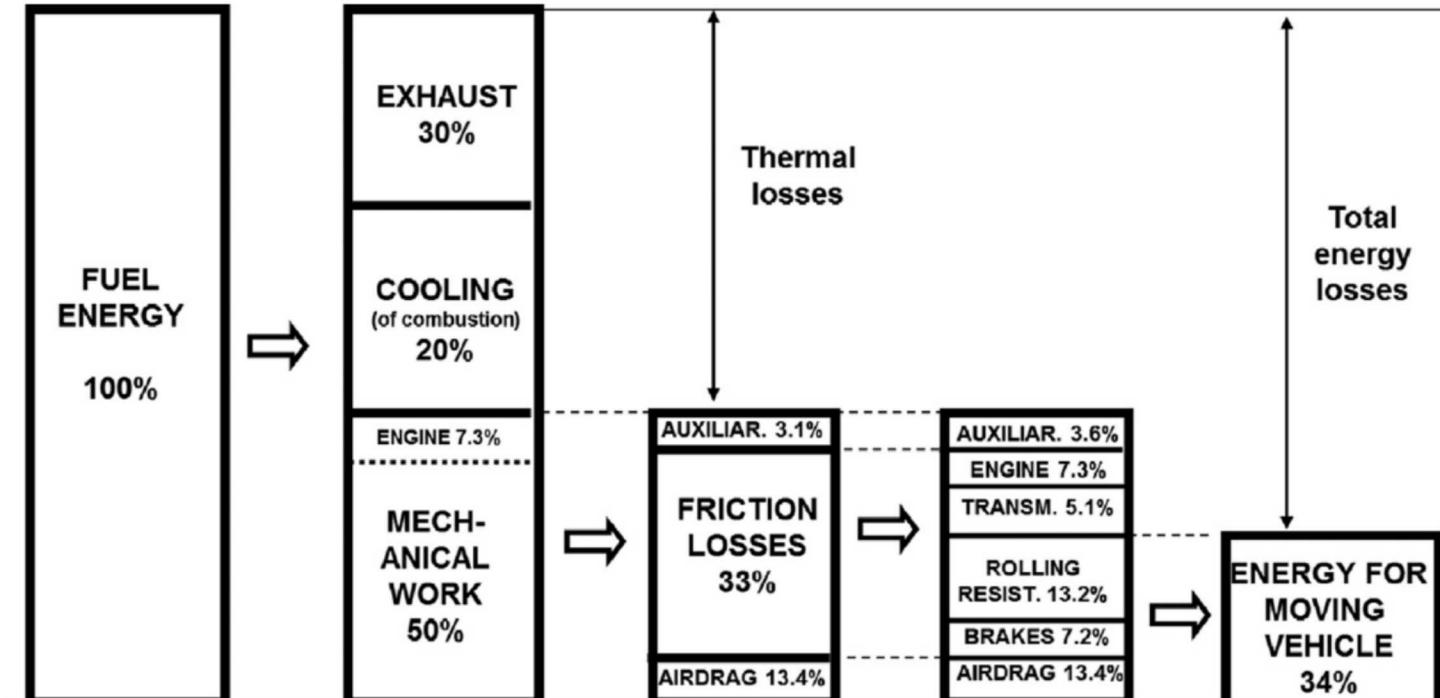
Mr. Justin Vanness
Senior Staff Member
Sandia National Labs

Special thanks to Dr. Michael Chandross
(who is typically a co-presenter...)



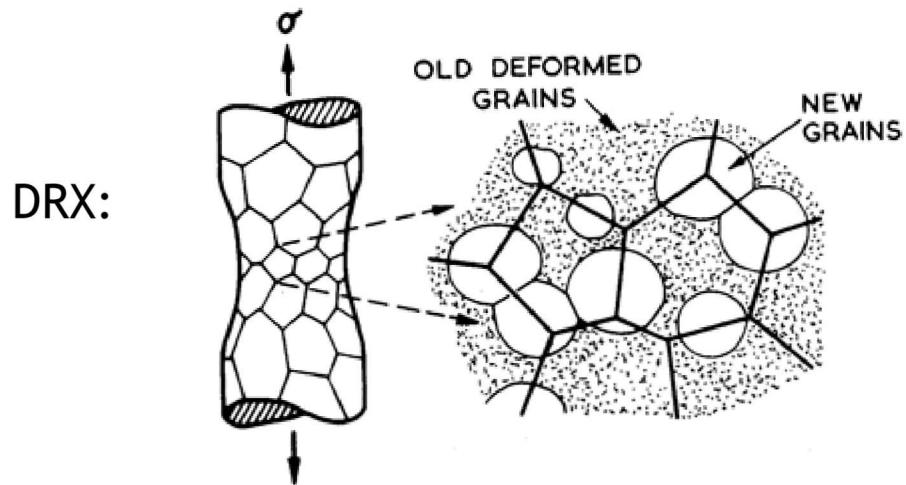
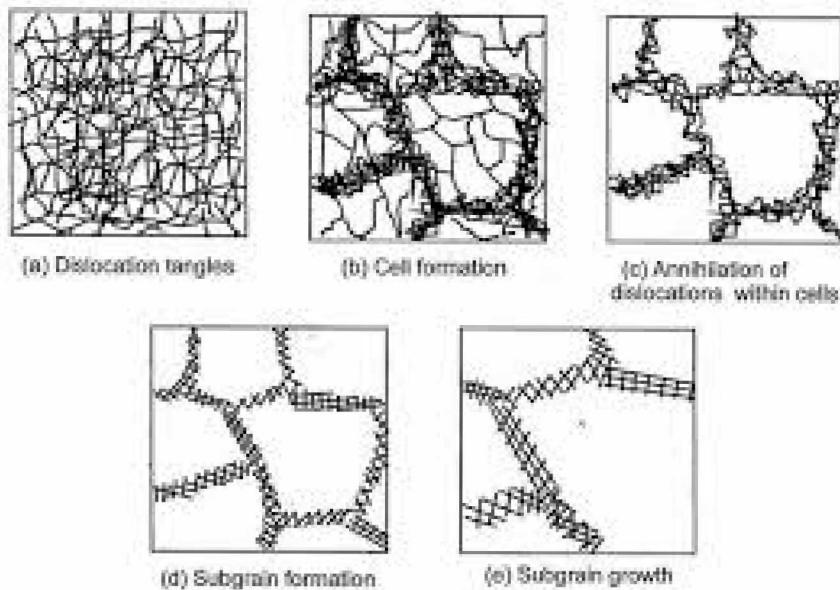
Questions?

Transportation – Passenger Car Energy Losses

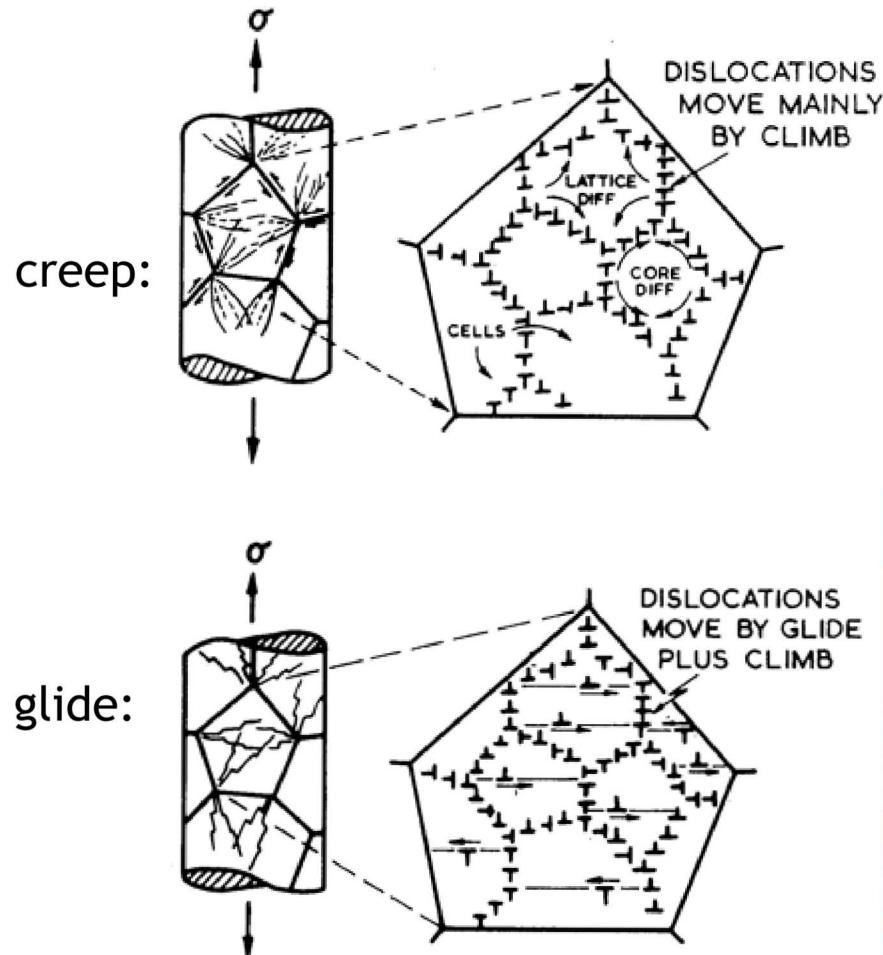


ref: K. Holmberg et al., *Tribo. Int.* 2012

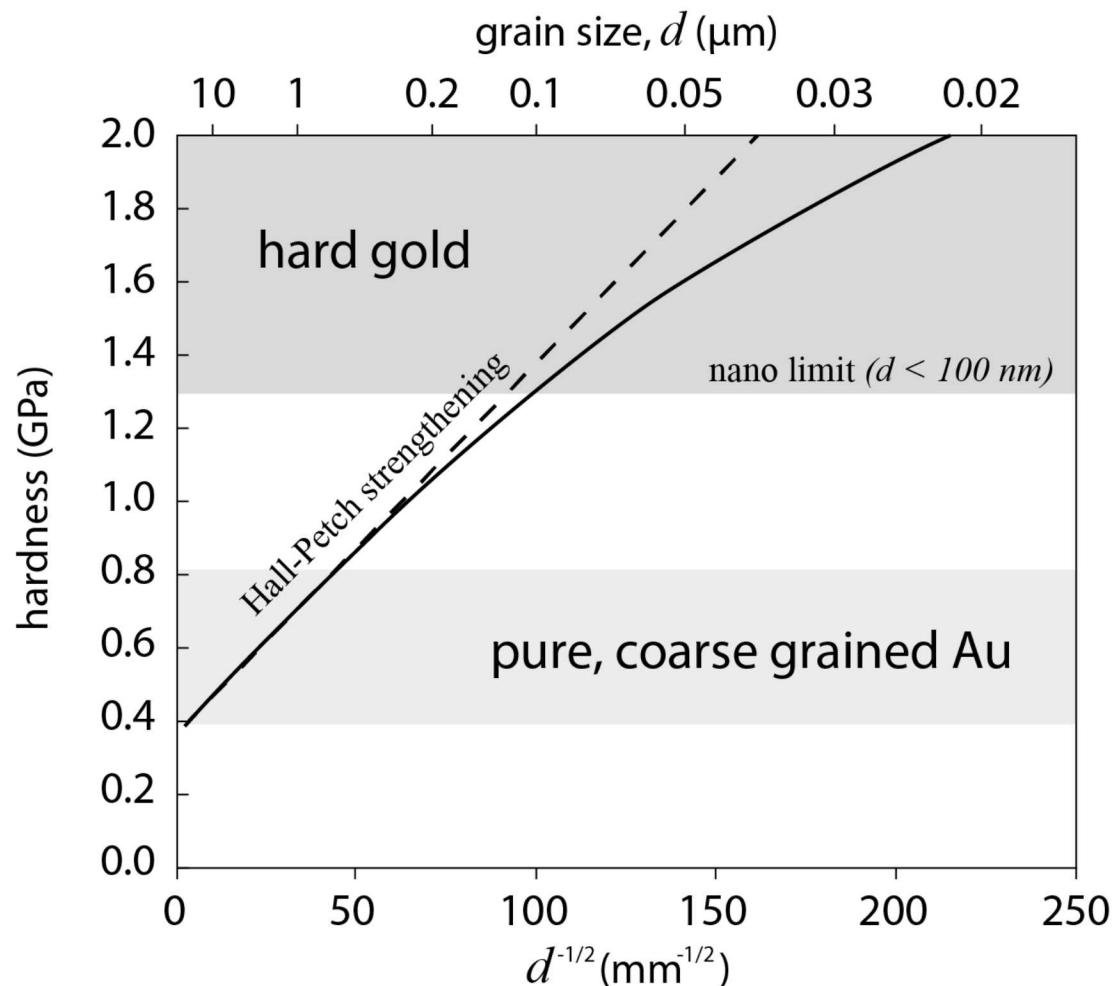
Competition between refinement and coarsening



http://engineering.dartmouth.edu/defmech/chapter_2.htm



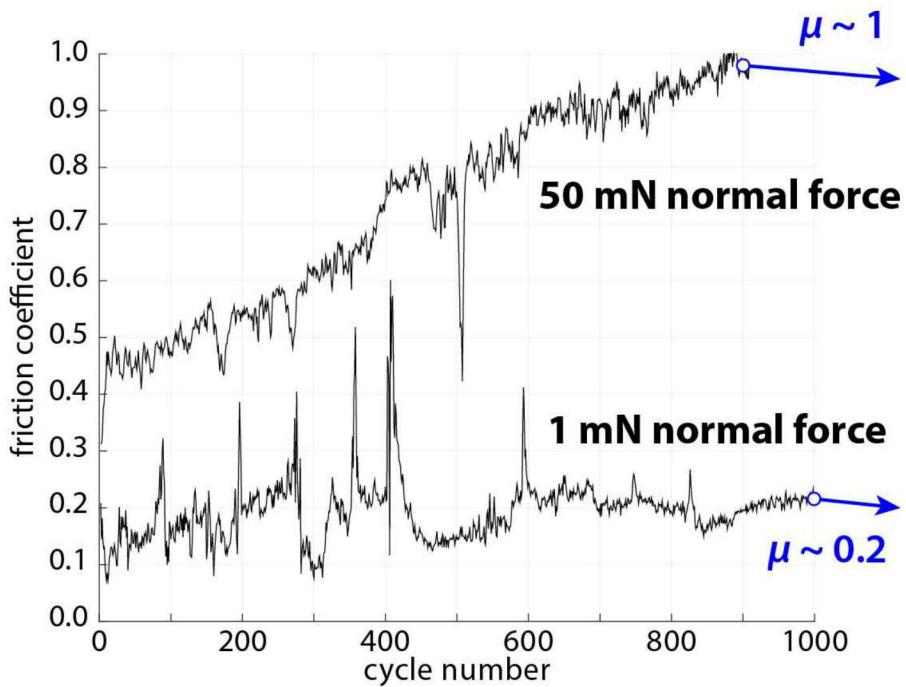
Grain-size dependent strength: Hall-Petch



Reference: C. Lo, J. Augis, and M. Pinnel, JAP (1979)

Electron diffraction of high and low friction wear tracks from Au-Au sliding contacts

Electron microscopy of focused ion beam prepared wear track cross-sections



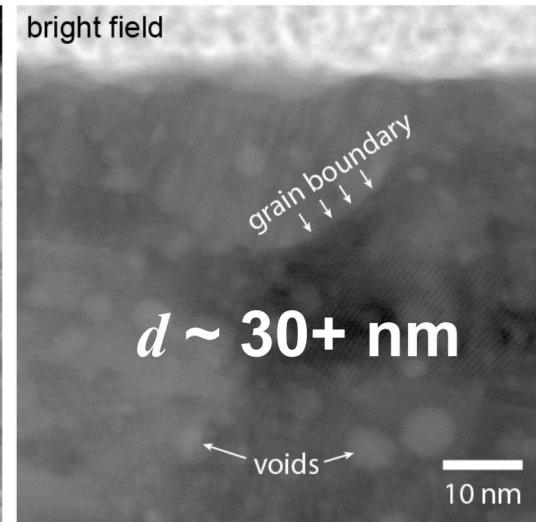
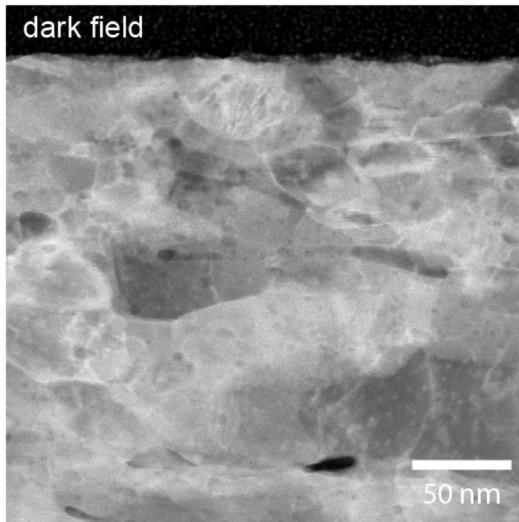
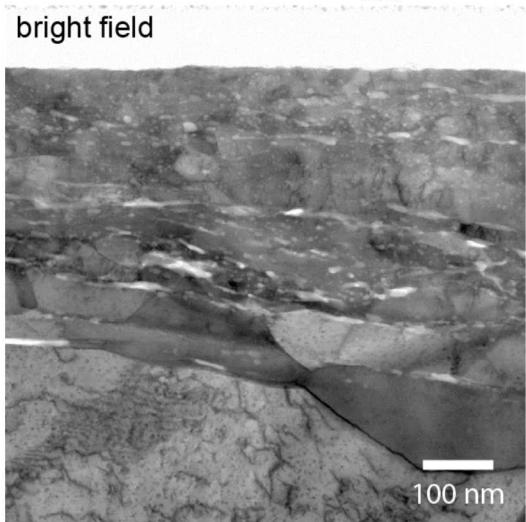
Transmission Kikuchi Diffraction (TKD):
(transmission diffraction performed in an SEM)



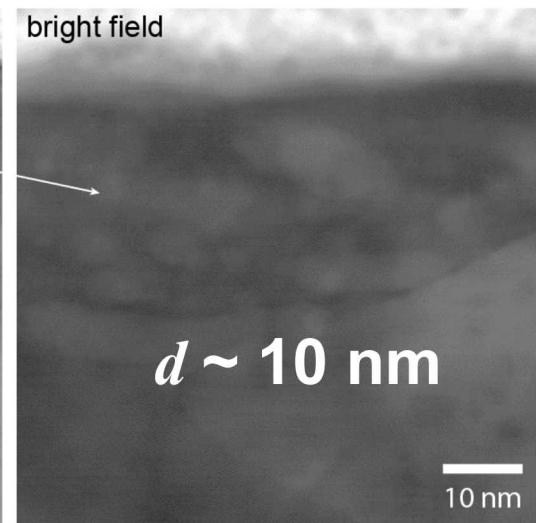
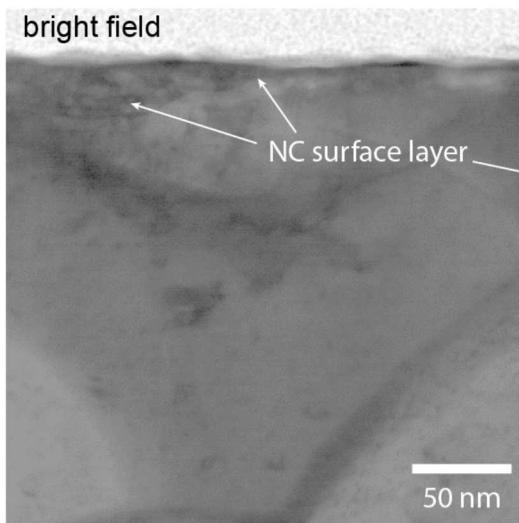
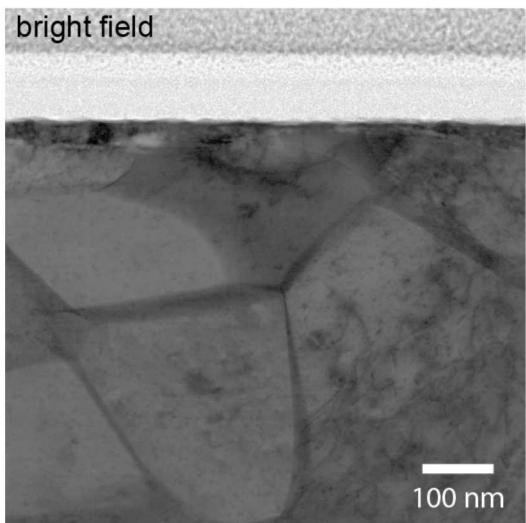
Again we see fine grain size in both cases...
but the low friction case seems smaller.

Microstructurally small differences in grain size = BIG difference in friction response

50 mN normal force, $\mu_{\text{final}} \sim 1.0$

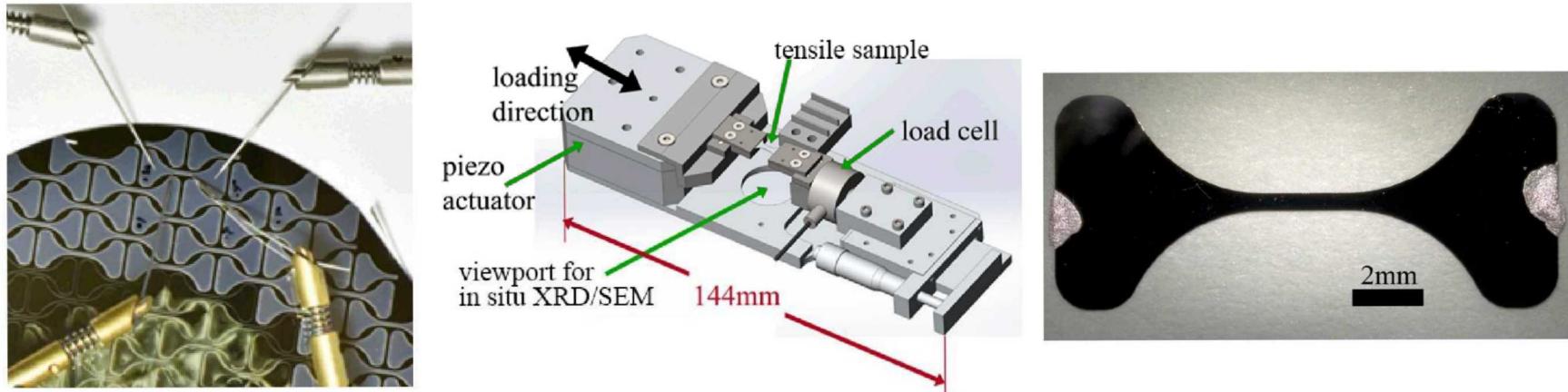


1 mN normal force, $\mu_{\text{ss}} \sim 0.2$

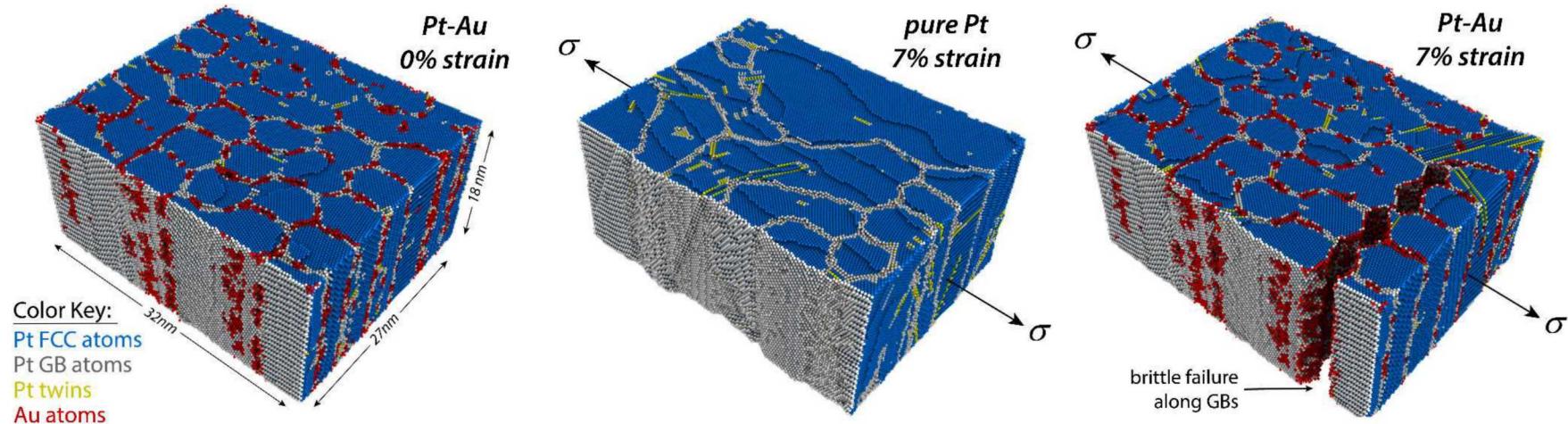


Tensile and fatigue properties of PtAu – Experiments & Simulations

Tensile Testing (Fatigue and Yield Strength):

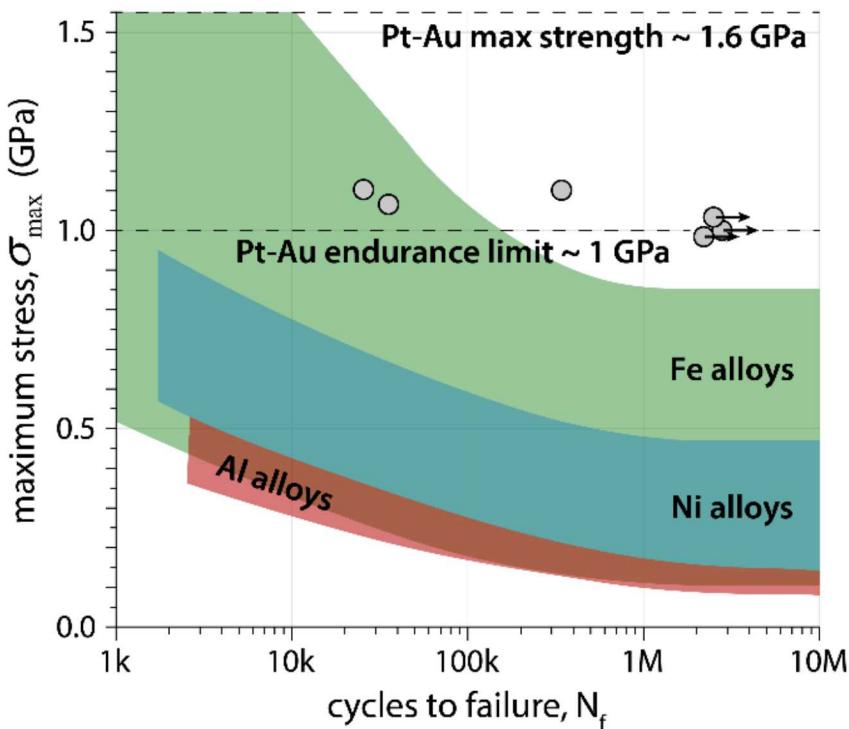


MD Simulation Snapshots (grain growth during uniaxial tensile stress exposure):

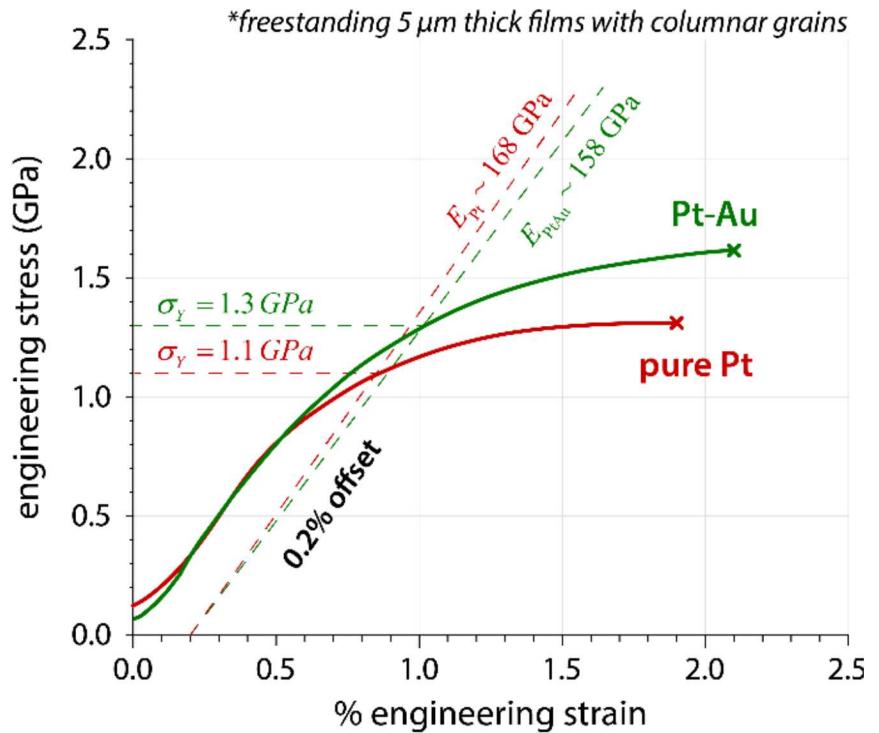


PtAu alloys have high yield strength and remarkable fatigue resistance

Experimental Tensile Fatigue:

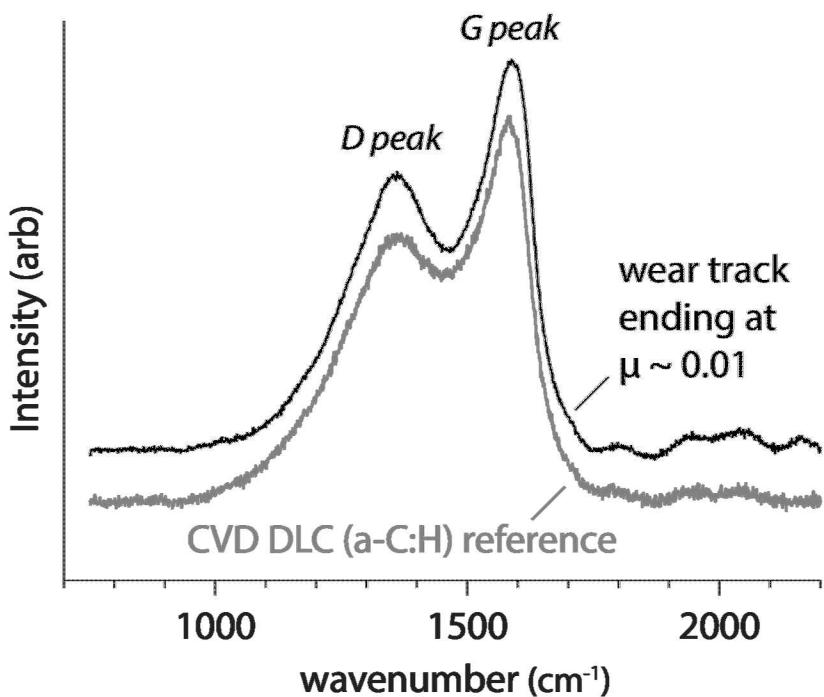


Experimental stress-strain data:

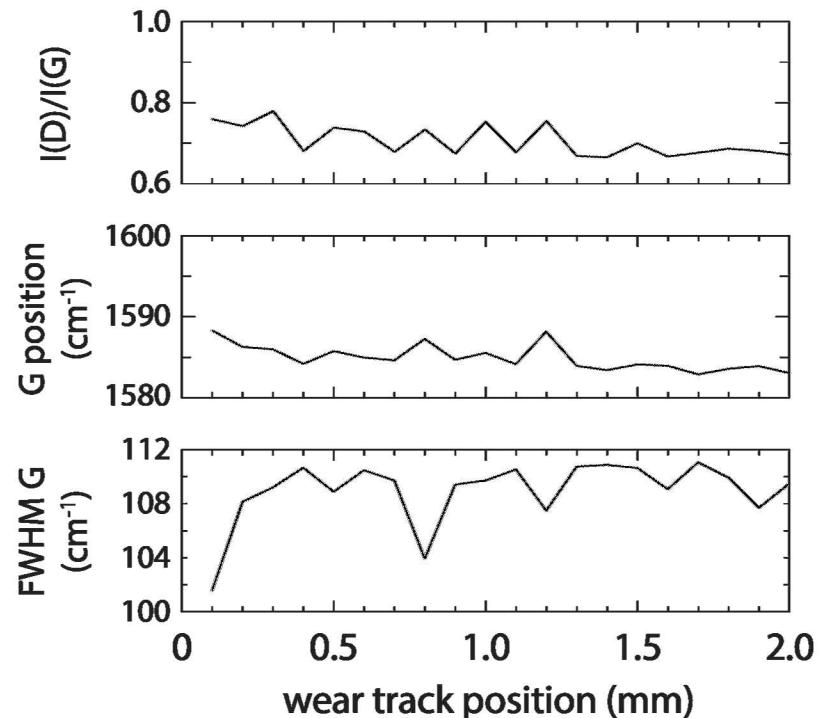


This high yield strength and fatigue resistance generated an interesting result...

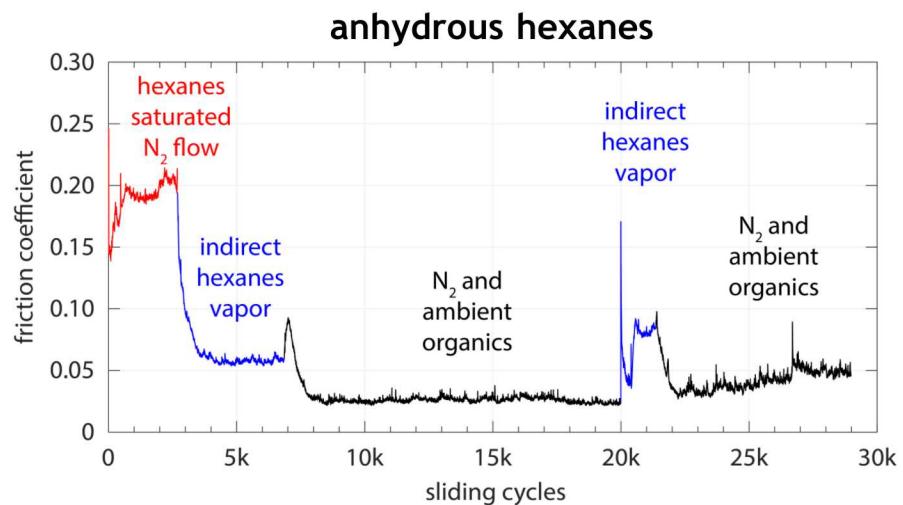
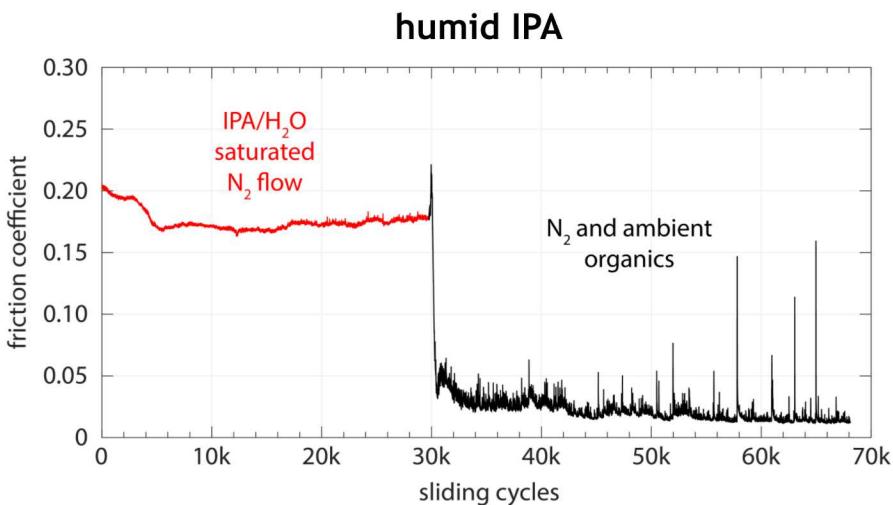
Raman Spectroscopy -- Highly Graphitic Hydrogenated (20%) DLC



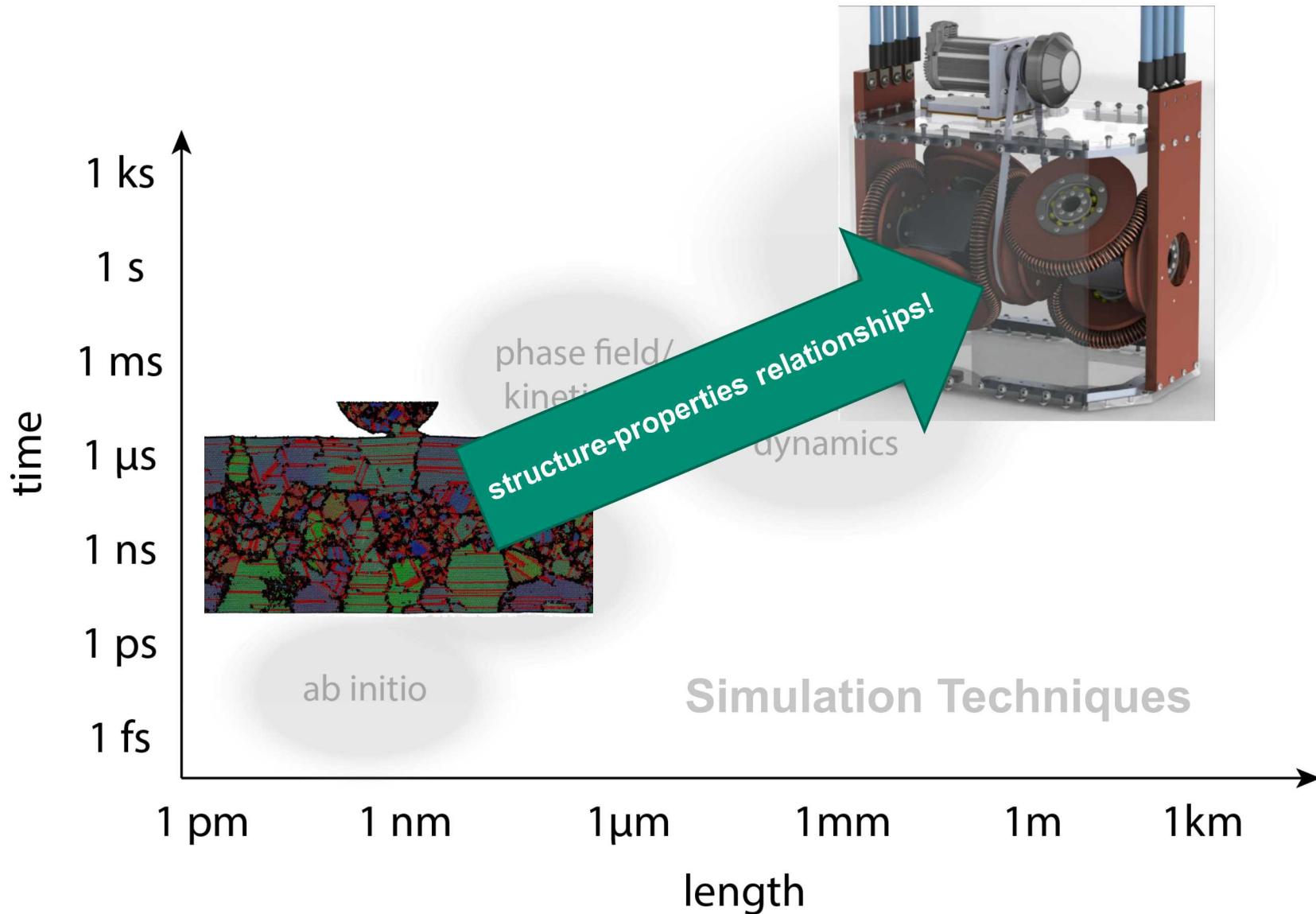
elastic recoil showed $\sim 20\%$ H



Friction behavior directly linked to hydrocarbon concentration (dry)

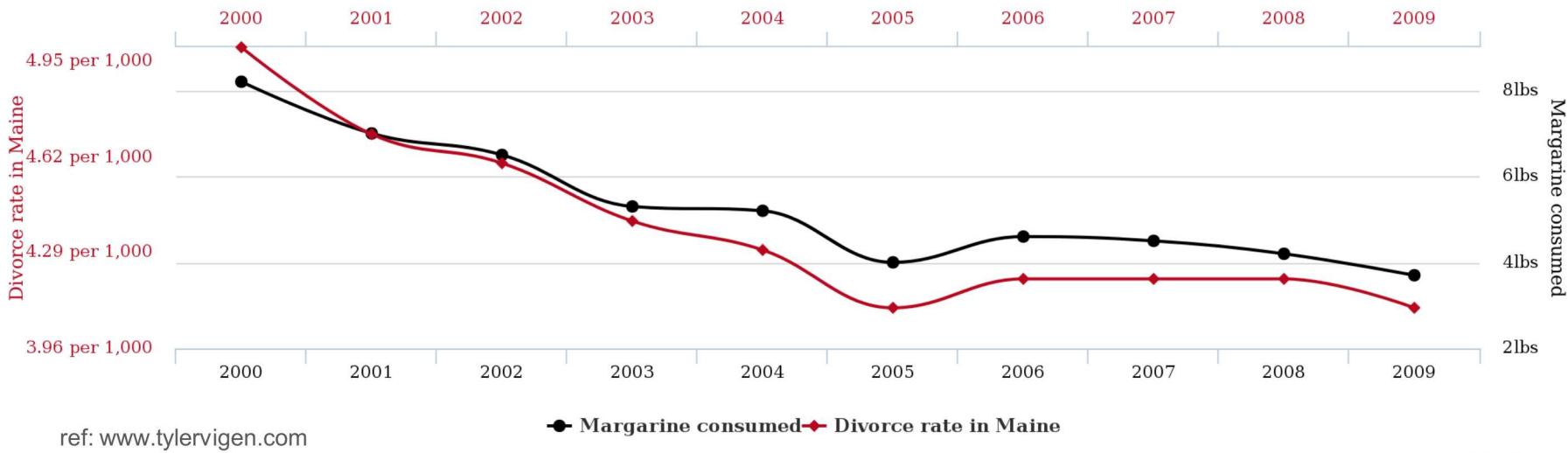


Alcohol helps, but only in moderation.



... as we know, correlation is not causation...

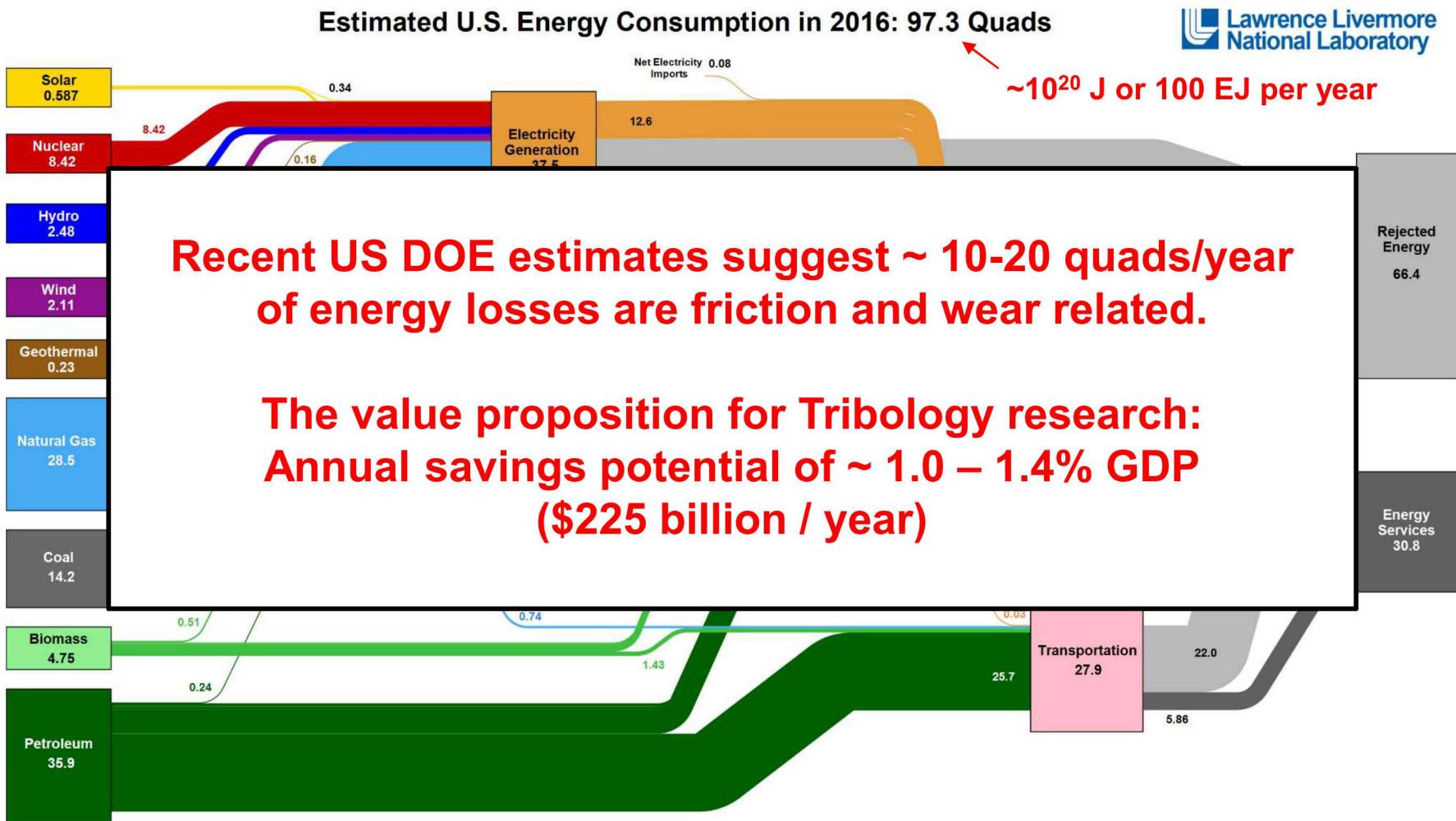
Divorce rate in Maine correlates with Per capita consumption of margarine



so please, for the sake of
married people in Maine...



Some Perspective on the Value Proposition of Tribology Research



Source: LLNL March, 2017. Data is based on DOE/EIA MER (2016). If this information or a reproduction of it is used, credit must be given to the Lawrence Livermore National Laboratory and the Department of Energy, under whose auspices the work was performed. This chart was revised in 2017 to reflect changes made in mid-2016 to the Energy Information Administration's analysis methodology and reporting. The efficiency of electricity production is calculated as the total retail electricity delivered divided by the primary energy input into electricity generation. End use efficiency is estimated as 65% for the residential sector, 65% for the commercial sector, 21% for the transportation sector, and 49% for the industrial sector which was updated in 2017 to reflect DOE's analysis of manufacturing. Totals may not equal sum of components due to independent rounding. LLNL-MI-410527

Attempts to define wear & friction regimes remain empirical/phenomenological

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et MATERIALIA

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VIEWPOINT SET No. 14

WEAR-MECHANISM MAPS

M. F. Ashby* and S. C. Lim+,

*Engineering Department, Cambridge University
+National University of Singapore, Kent Ridge

(Received August 15

(Revised October 16

WEAR-MECHANISM MAPPING:

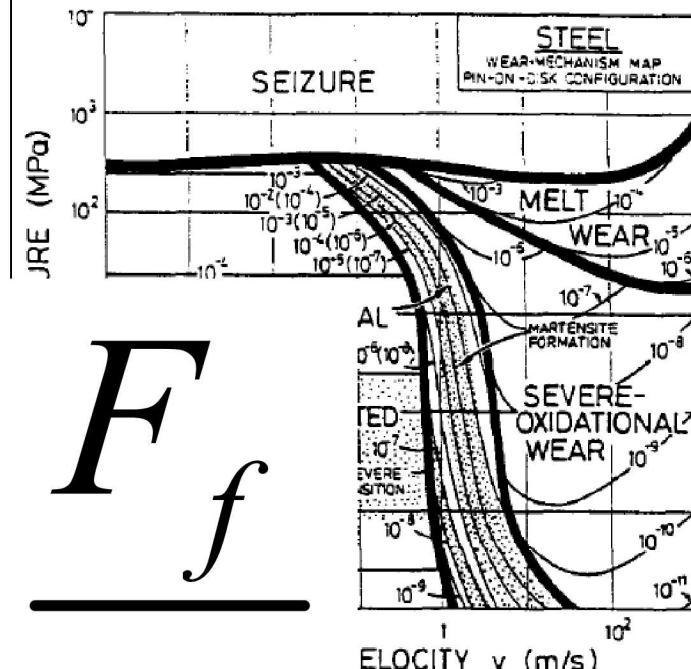
Wear is the loss or transfer of material in general, the wear rate W (defined here as the volume of material removed per unit distance slid) depends on the bearing pressure F carried by the contact and A_n is its nominal area and on the material properties and geometry of

$$W = f(F/A_n, v, \text{Mat. Props., Geomet.})$$

But one such equation is not enough. There are different mechanisms of wear, dependent in a different way on the variables. F and v , is the one leading to the fastest rate of wear, mechanisms encountered in wear studies of metal melting, by chemical change induced by friction, plasticity and by brittle fracture.

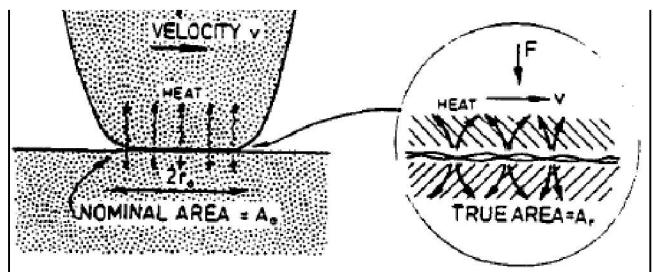
TABLE 1: MECHANISMS OF

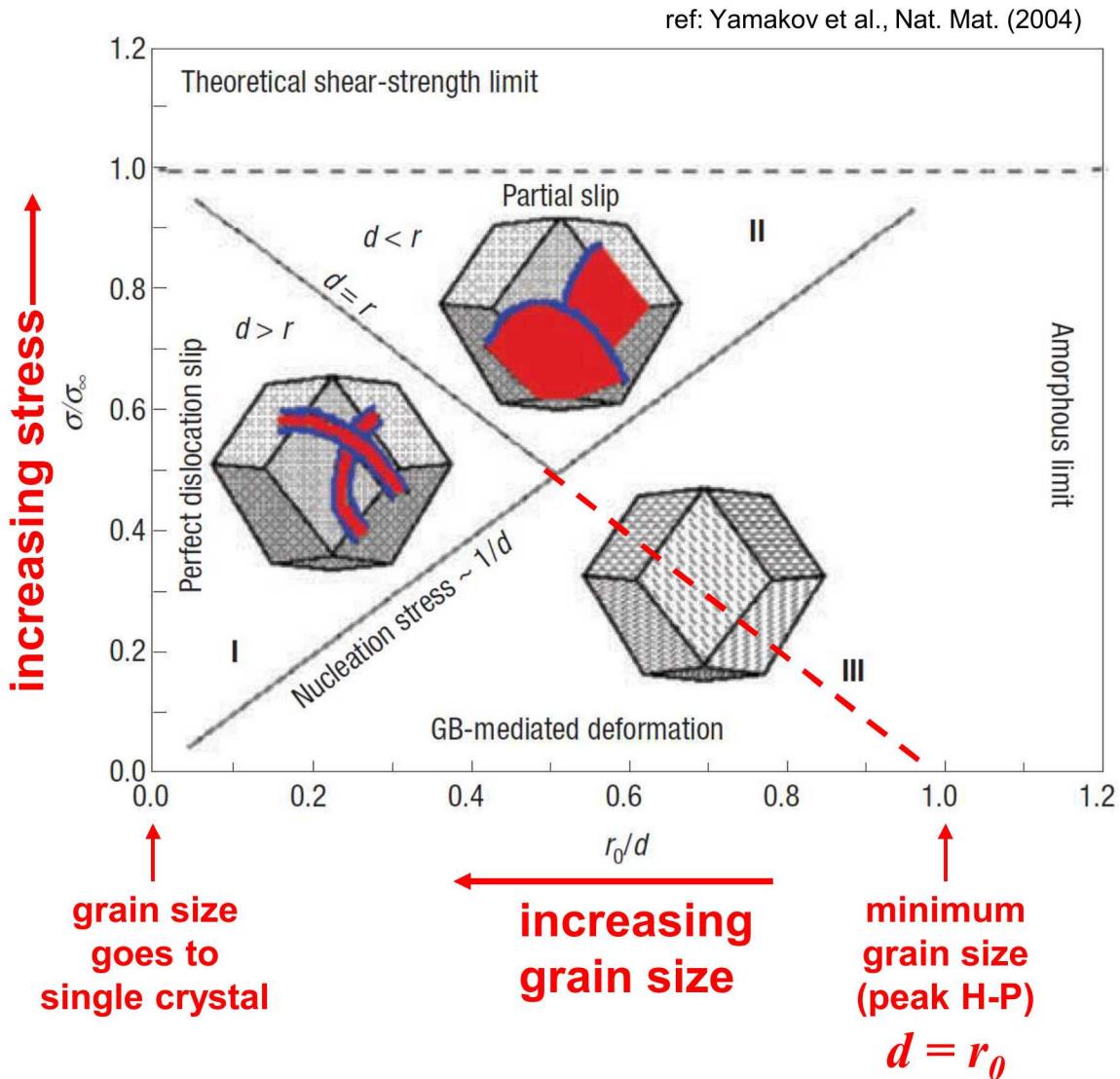
METALS	CERAMICS
SEIZURE	SEIZURE (
MELT WEAR	MELT WEAR
SEVERE-OXIDATIONAL WEAR	THERMALLY
MILD-OXIDATIONAL WEAR	THERMAL C
PLASTICITY-DOMINATED WEAR	BRITTLE S
ULTRA MILD WEAR	TRIBOCHEM



mechanism map for low-
velocity wear. It is based on physical modelling
of the wear process. The shaded regions
indicate specific physical modelling regimes.

$$\mu = \frac{F_f}{F_n}$$





Equilibrium (zero stress) dislocation splitting distance:

$$r_0 = \frac{(2 + \nu) G b^2}{4\pi(1 - \nu) \gamma_{sf}}$$

Stress-dependent splitting distance:

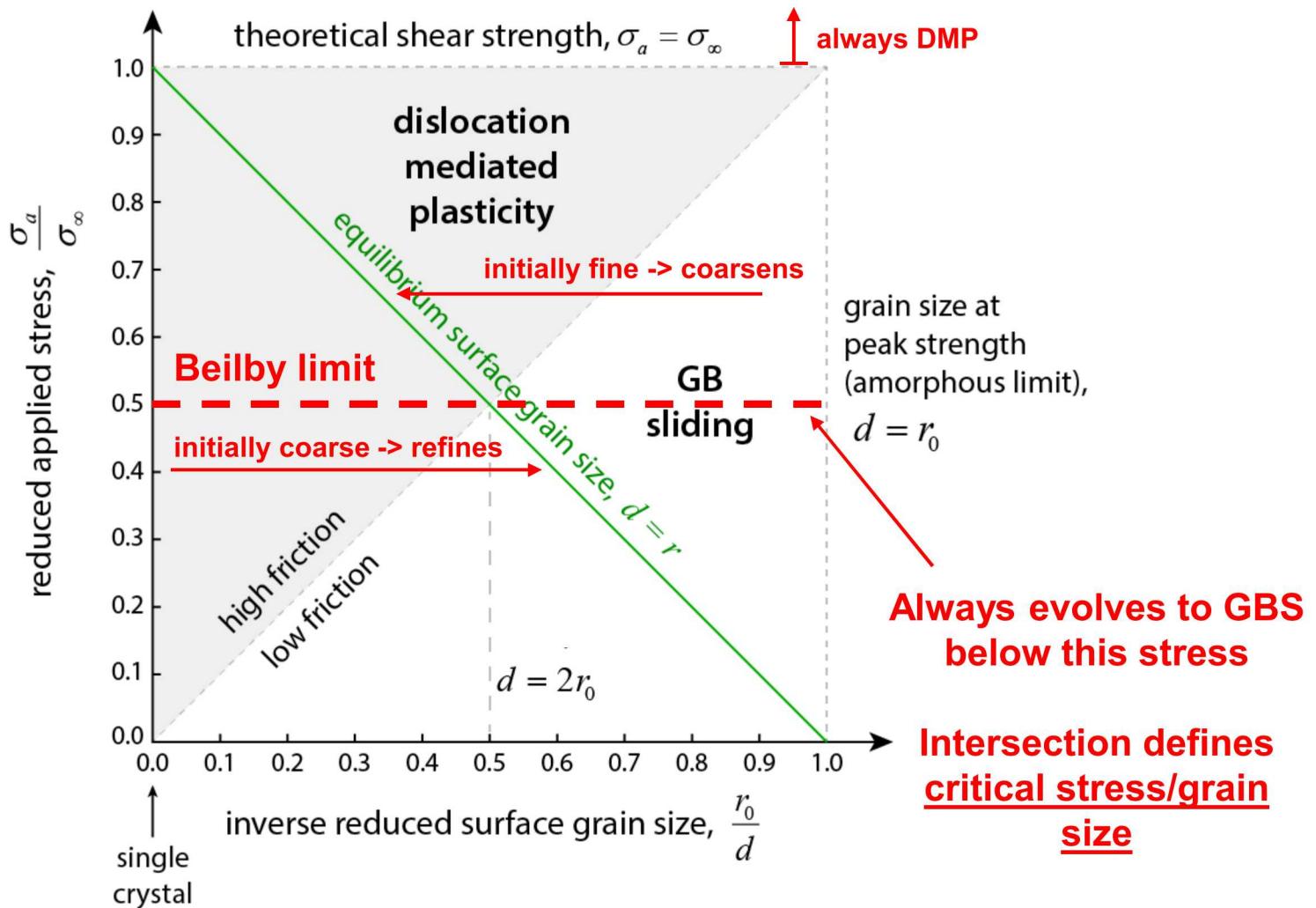
$$r = \frac{r_0}{1 - \sigma_a / \sigma_\infty}$$

Theoretical shear strength:

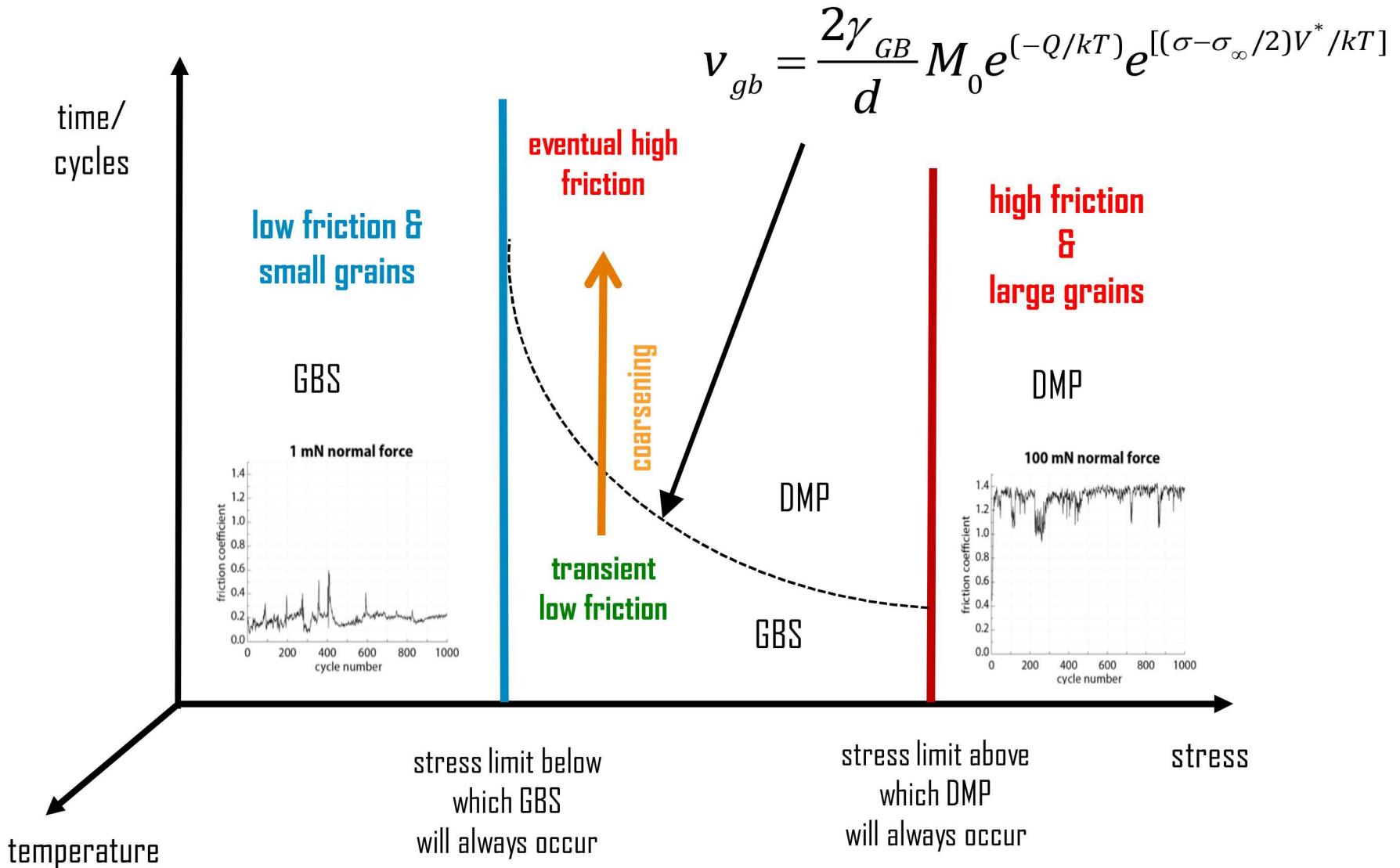
$$\sigma_\infty = \frac{2\gamma_{sf}}{b}$$

Ref: Froseth et al., Acta Mat. (2004)

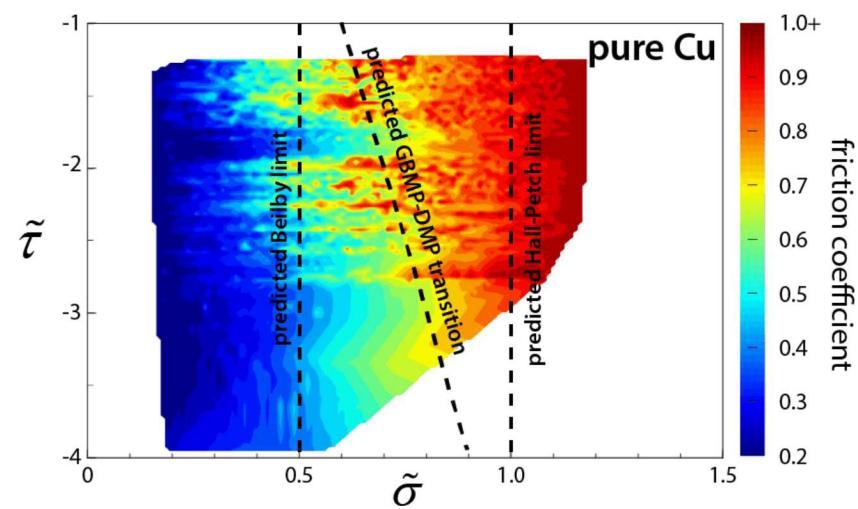
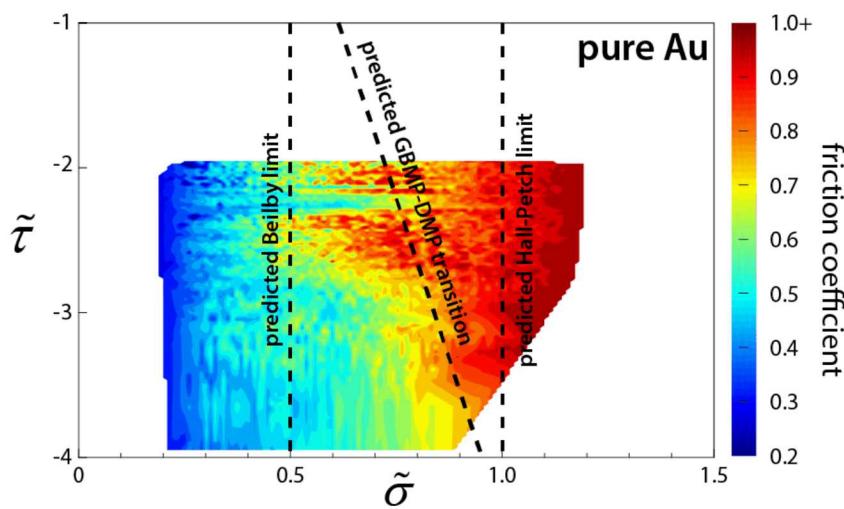
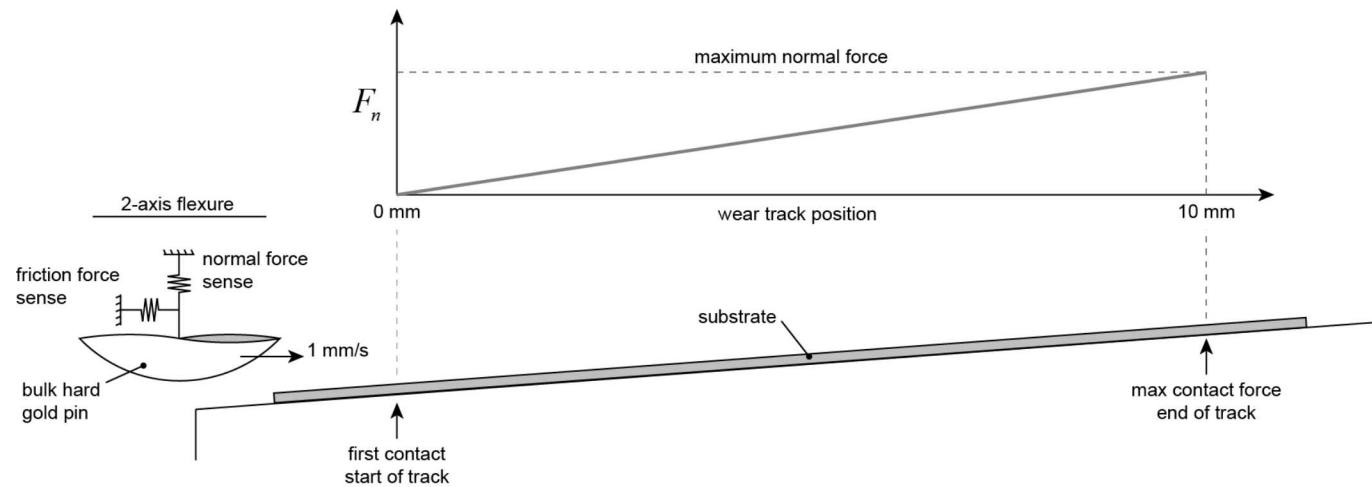
Recasting the Yamakov et al. criteria for tribology (& time-dep)



Generalized friction regimes map for metals

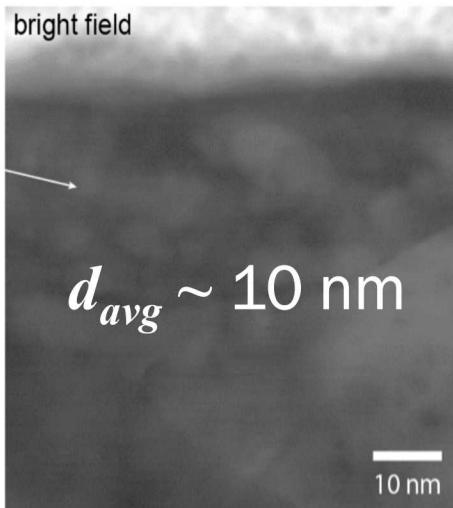


How can we test theory in one shot? Ramped contact force experiments

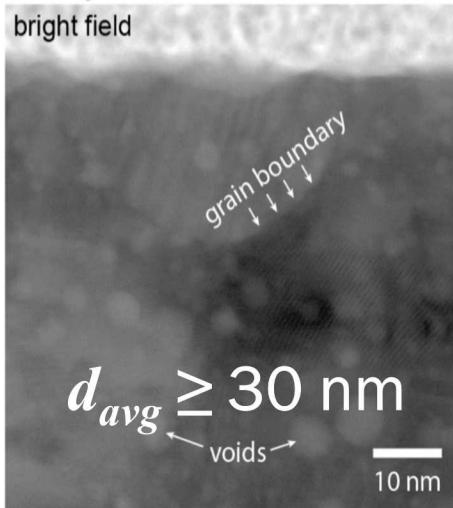


We can predict a crossover critical grain size

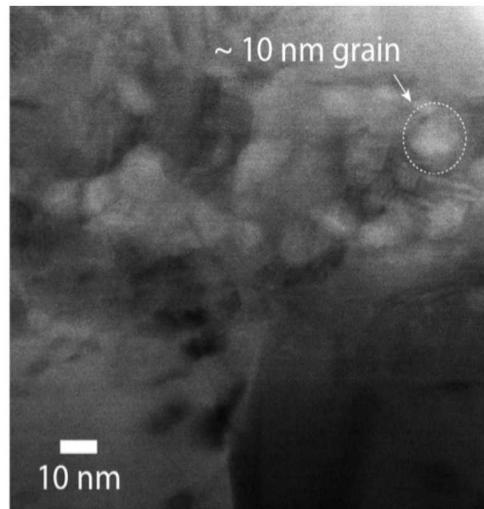
gold on gold



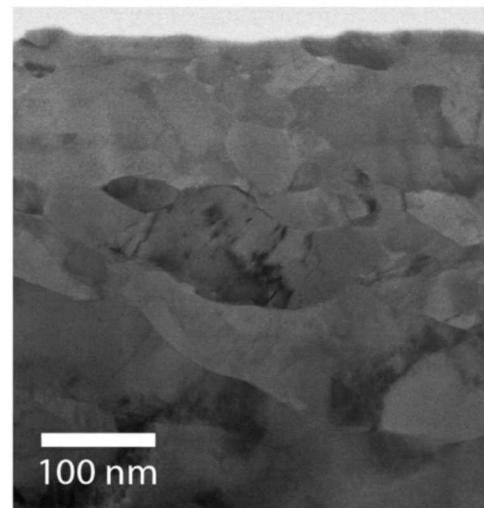
critical grain size = 17.4 nm



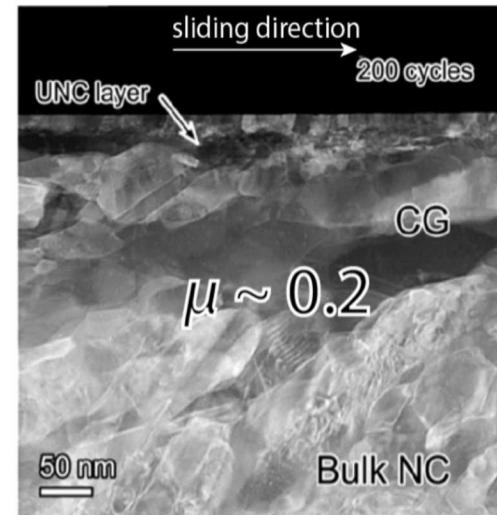
copper on copper



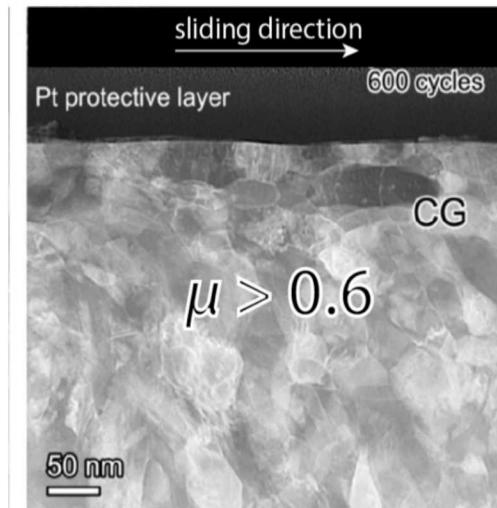
critical grain size = 11.8 nm



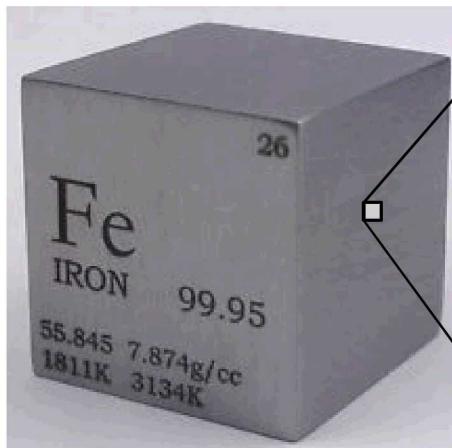
Si_3N_4 on nickel



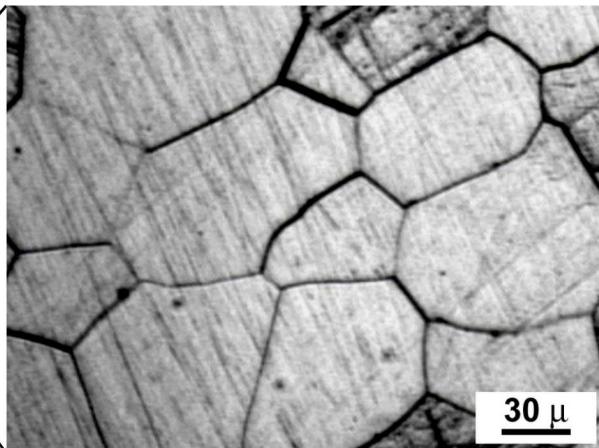
critical grain size = 9.8 nm



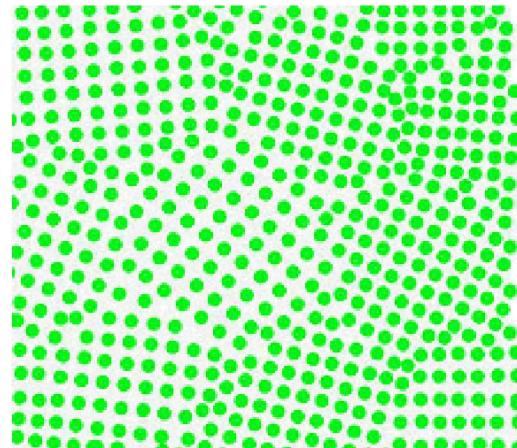
Crystal Structure of Metals (A Primer)



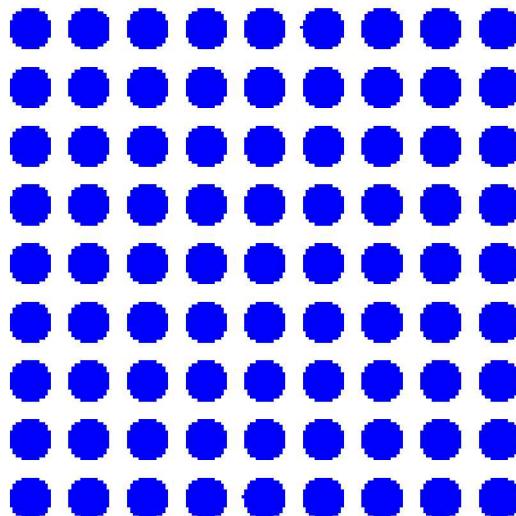
<https://www.amazon.com/25-4mm-Metal-99-95-Engraved-Periodic/dp/B0BWVQ7JX0>



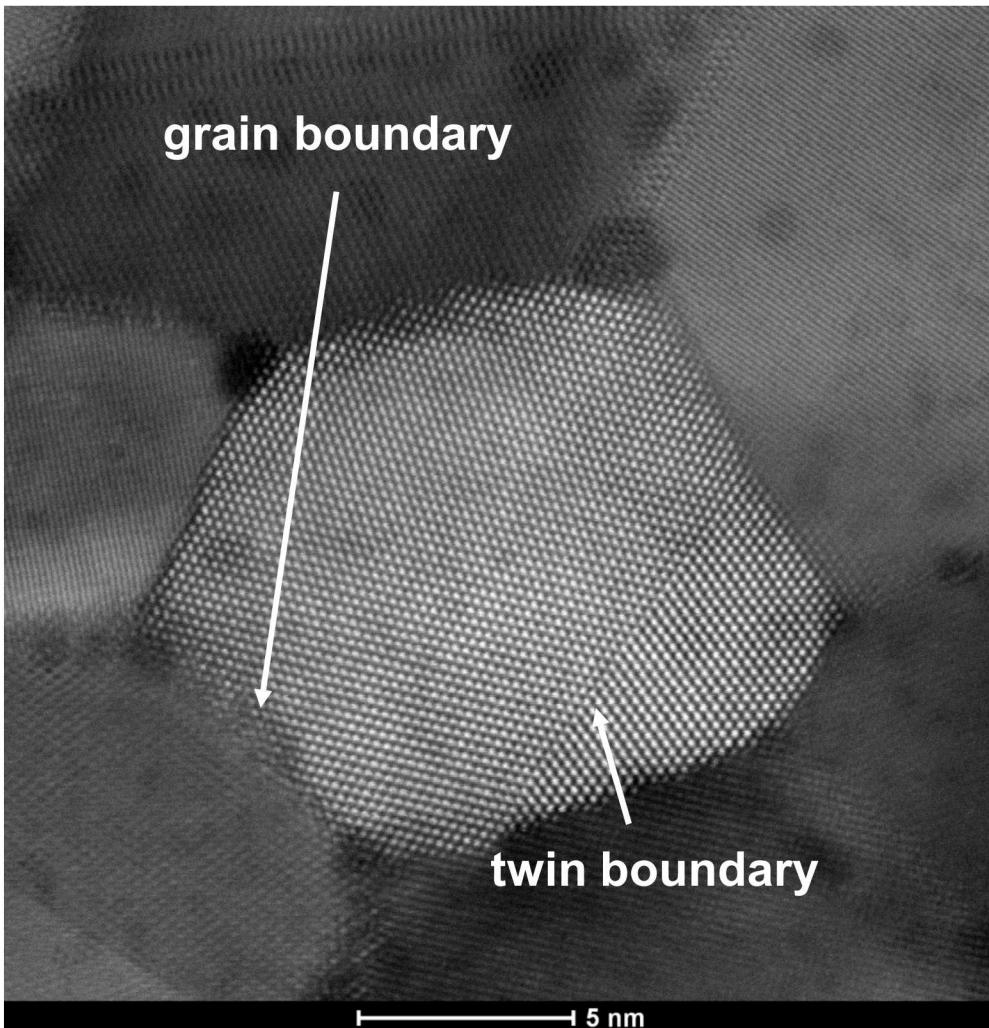
https://en.wikipedia.org/wiki/Grain_boundary



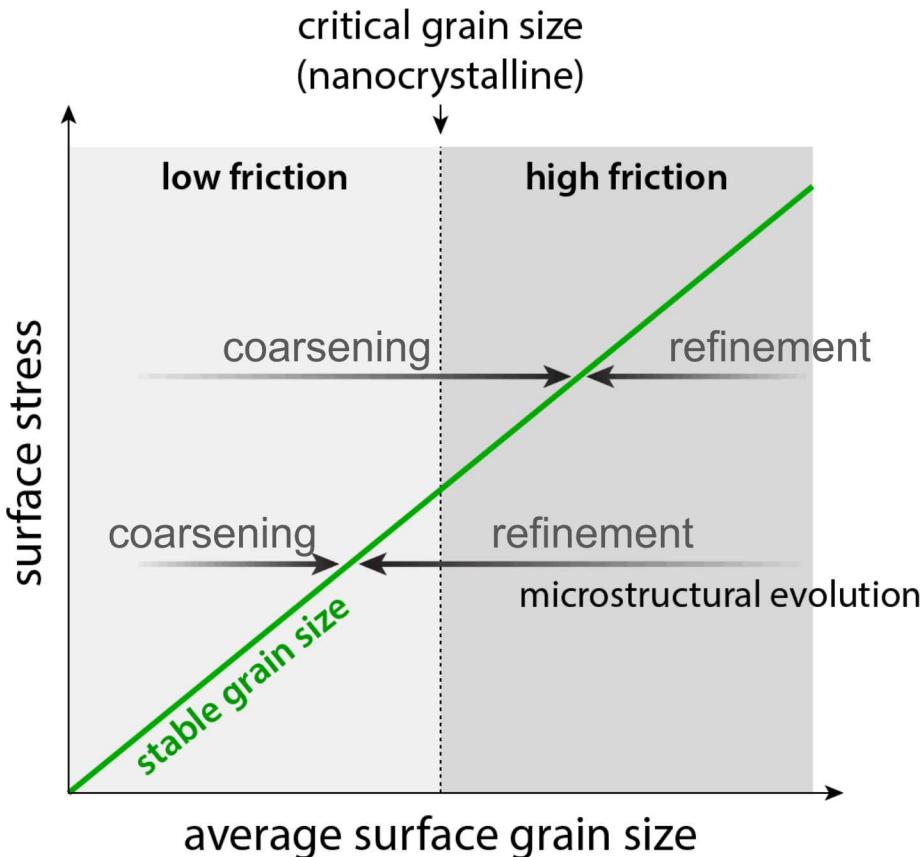
Shear via
dislocation
motion:



"Crystals are like people, it's the defects that make them interesting."
- C. J. Humphreys



Grain size evolution is a competition (growth/refinement)



“All models are wrong, but some are useful.”
--George Box

“It is better to be vaguely right than exactly wrong.”
--Carveth Read

- Effective refinement from recrystallization (Zener & Holloman, 1944; McQueen et al., 1967)
- Known in rocks and ice cores (Derby et al., 1992)
- Recently extended to metals under severe plastic deformation (Pougis et al., 2014)