

The Critical Microstructural Conditions Leading to the Formation of Voids During Ductile Rupture



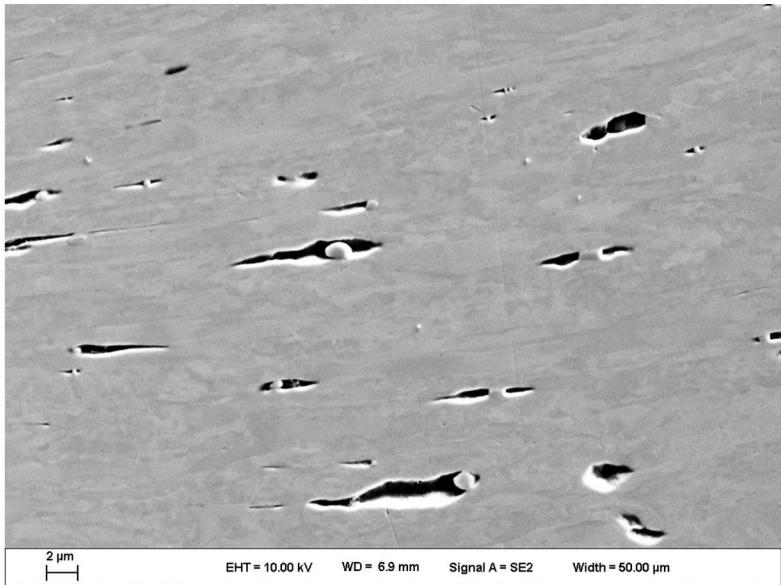
Philip J. Noell, Julian E. C. Sabisch, Douglas L. Medlin, Jay D. Carroll, Brad L. Boyce



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Void nucleation during ductile rupture

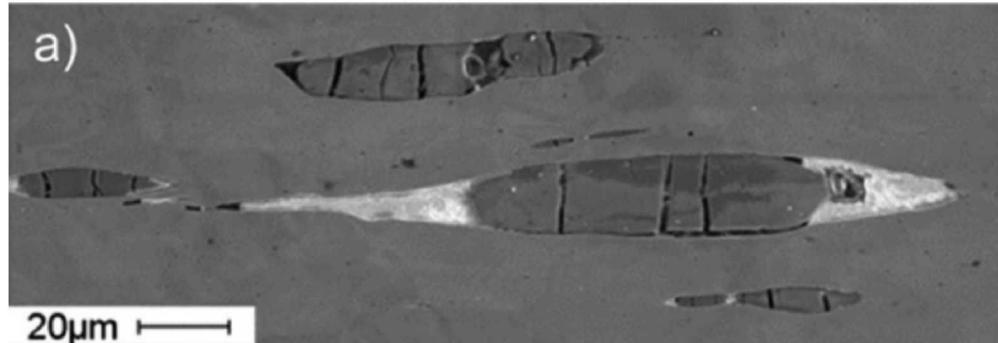
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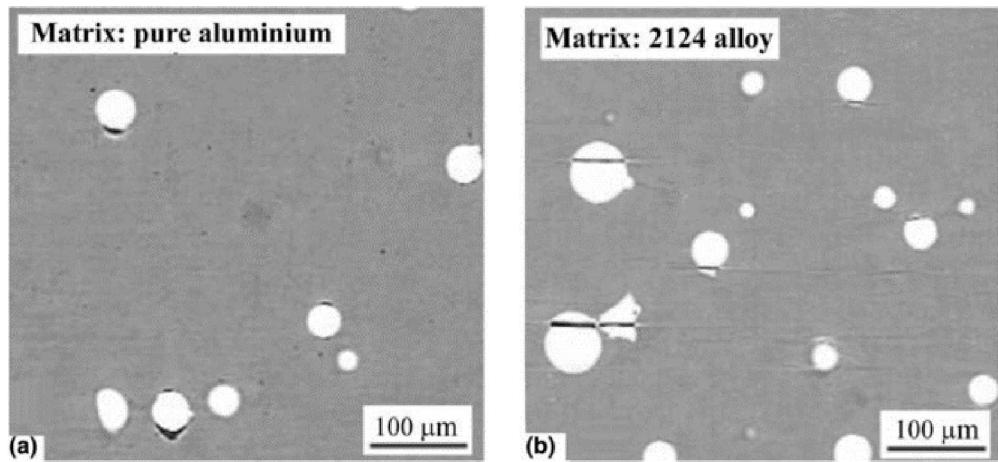
Secondary-electron image of voids in a deformed copper wire containing second-phase particles

Void nucleation in many materials is intragranular and associated with inclusions and/or second-phase particles, either by particle fracture or by particle decohesion.

Does microstructural evolution in the matrix during deformation play a role in void nucleation?



Cracked MnS inclusions in a free-cutting steel. Seo et al, ISIJ International, 2015.



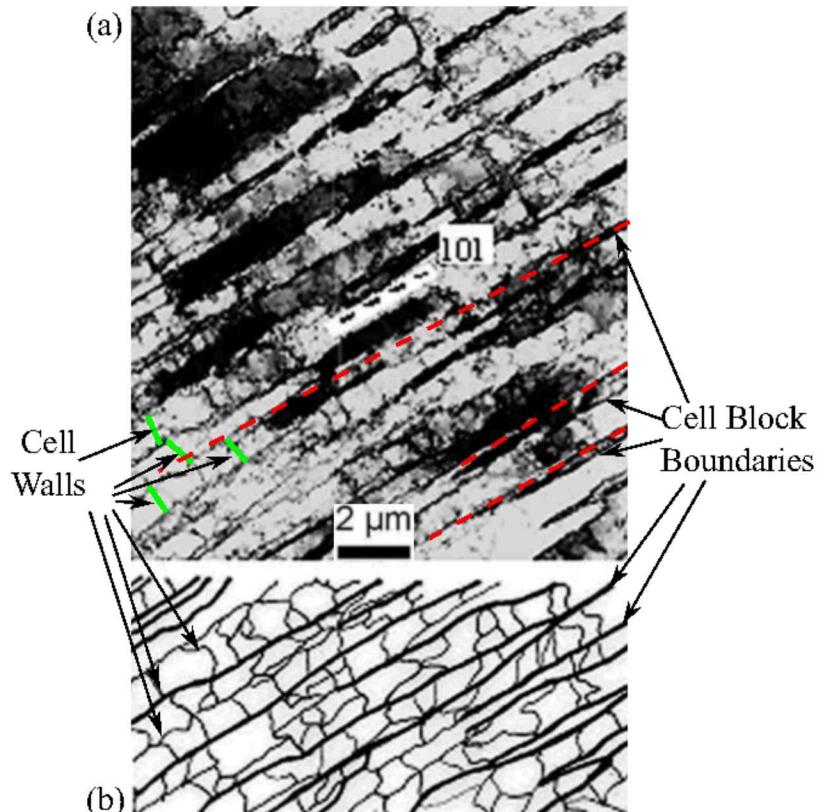
Babout, L. et al, *Acta Mater.*, (2004)

Deformation-induced defect structures

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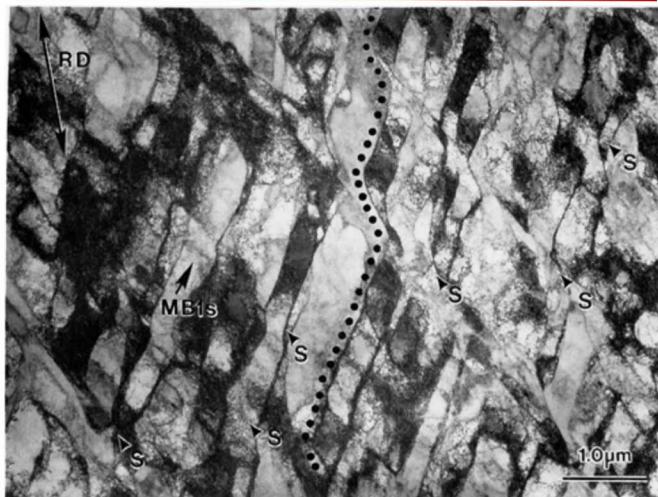
- Deformation twins
- Slip bands and dislocation pileups
- Dislocation boundaries
 - cell block boundaries, cell walls, microshear bands
- Dislocation loops, stacking fault tetrahedra
- Vacancy clusters

Cell blocks and cell walls in deformed iron



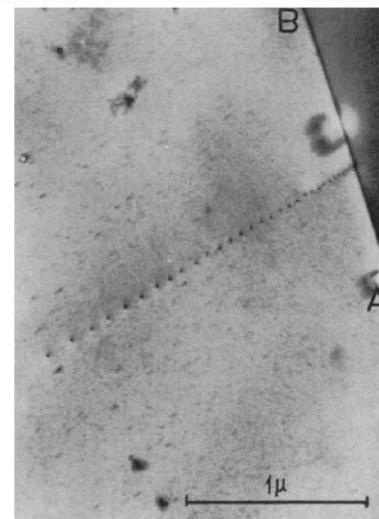
Li, B. L., et al. *Acta Materialia* 52.4 (2004)

A microshear band cutting through cell blocks in Ni



Hughes, D.A. et al., *Met Trans A*, 1993

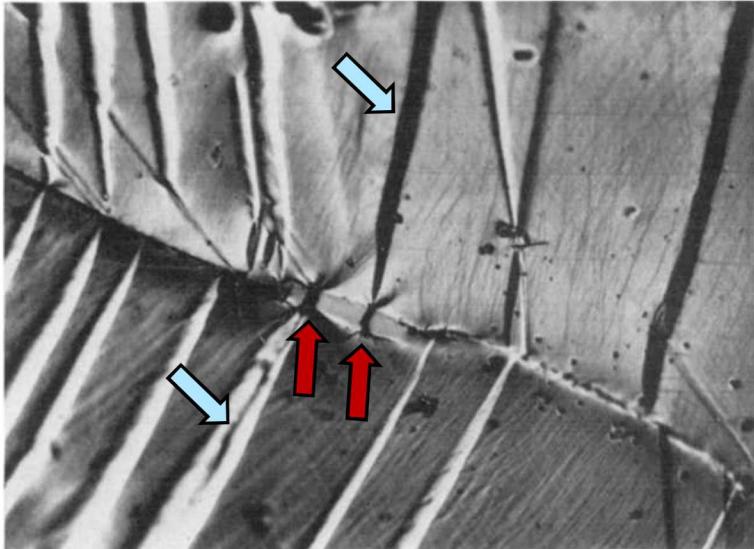
A dislocation pile-up at a grain boundary in tungsten



Wronski, A. et al., *J. Less Comm. Metl*, 1964

Relating particle-stimulated void nucleation to deformation-induced defect structures: dislocation pileups and deformation twins

Fractured hydride platelet in Zr associated with $\{11\bar{2}1\}$ type twins; **particle fracture depended on twin type**

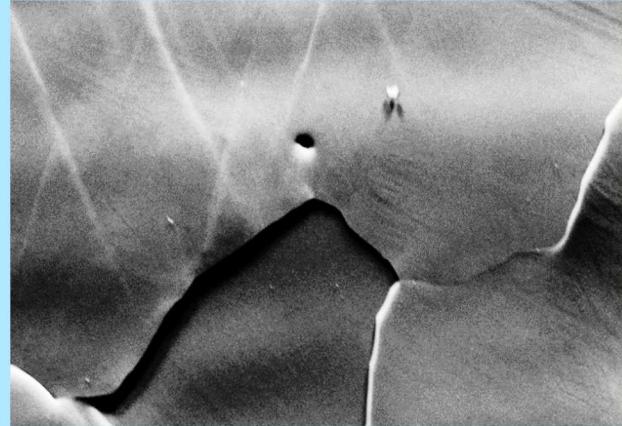


Warren, M. et al., *J. Nuc. Mat.*, 1968

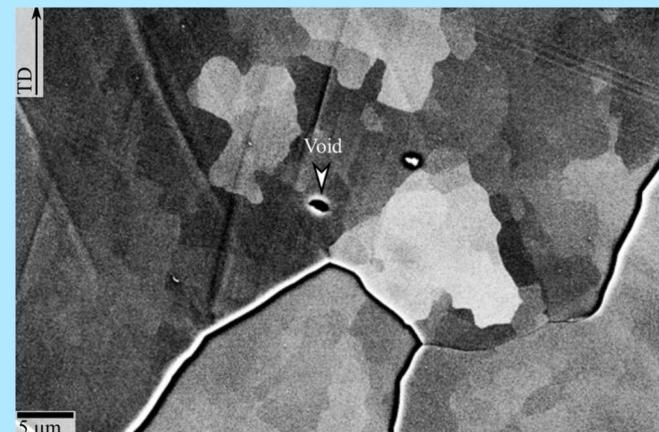
- Early studies of void nucleation based on secondary electron images, overlooked the role of dislocation boundaries in void nucleation
- Modern X-ray tomography studies of void nucleation no insight into relationship between void nucleation and defect structures

A cracked carbide associated with slip bands in an austenitic stainless steel

SE Image: only slip bands visible

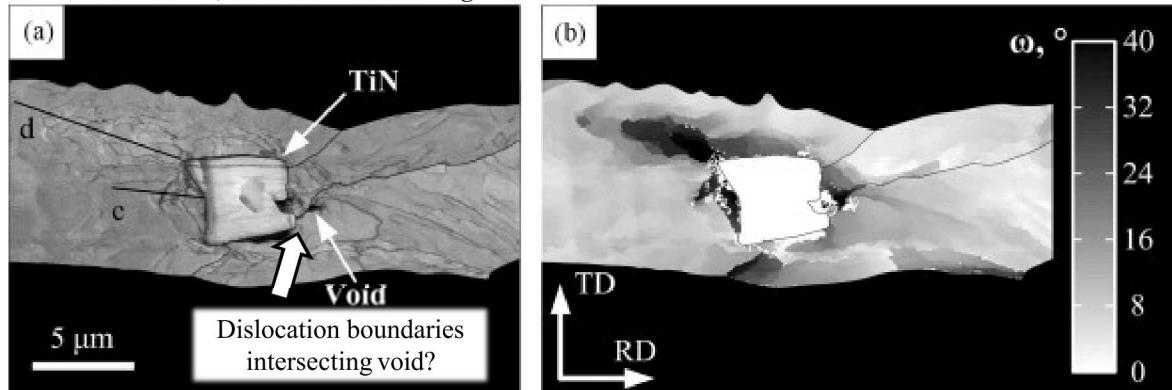


BSE Image needed to observe dislocation boundaries



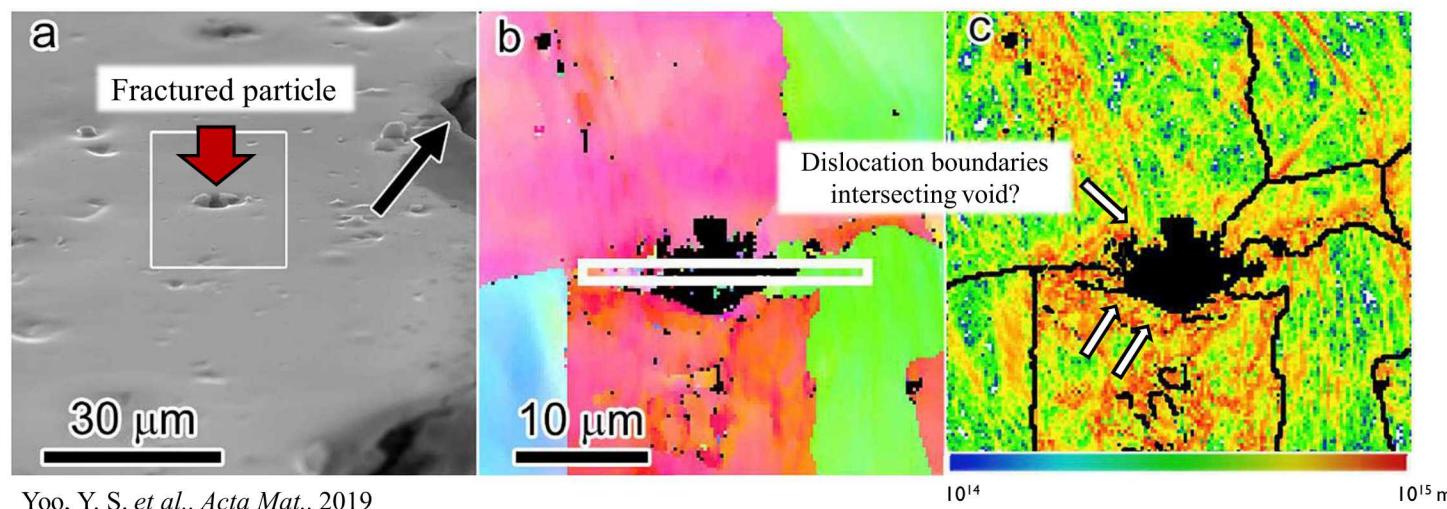
Relating particle-stimulated void nucleation to deformation-induced defect structures: dislocation boundaries

Leon-Garcia, O. et al., *Mat. Sci. Eng. A.*, 2010



Particle debonding at a TiN precipitate in an IF steel; EBSD data suggest that debonding was associated with a dislocation boundary, but data resolution is poor

Fracture of an intermetallic particle in Al6061, EBSD data suggest high density of dislocation boundaries intersected fractured particle



Yoo, Y. S. et al., *Acta Mat.*, 2019

Two prior studies contain examples of particle debonding and fracture near high-angle dislocation boundaries, but *no detailed characterization of the relationship between incipient voids and deformation defects was performed*

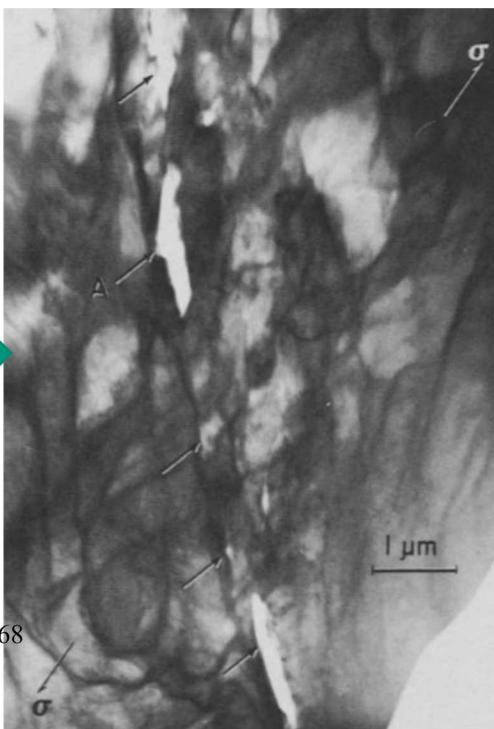
The microstructural features associated with particle-free void nucleation in single crystals

(a) The dislocation structure in a strained beryllium (HCP) single crystal



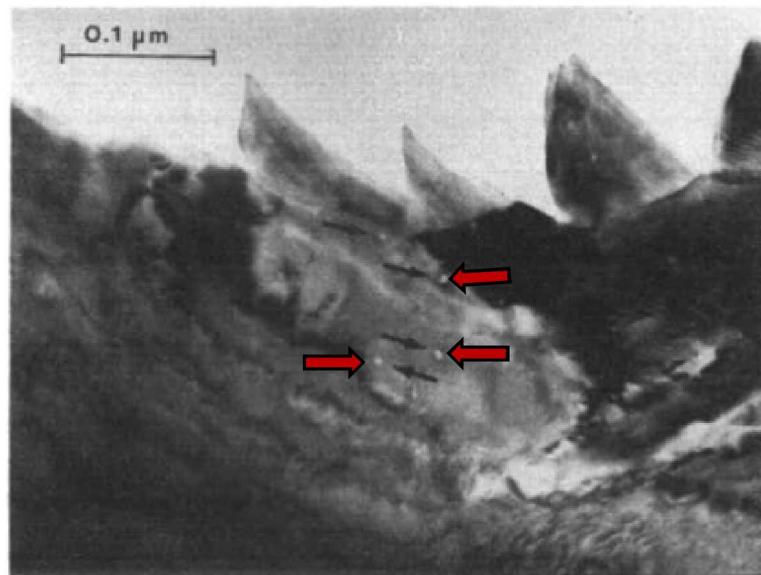
Warren, M. et al., *J. Nuc. Mat.*, 1968

(b) The same region as in (a) after further straining. Voids along the dislocation boundary indicated with black arrows



Gardner, R.N., et al. "Crack initiation at dislocation cell boundaries in the ductile fracture of metals." *Materials Science and Engineering* 29.2 (1977): 169-174.

Vacancy clusters near the fracture surface of a single-crystalline, gold thin film

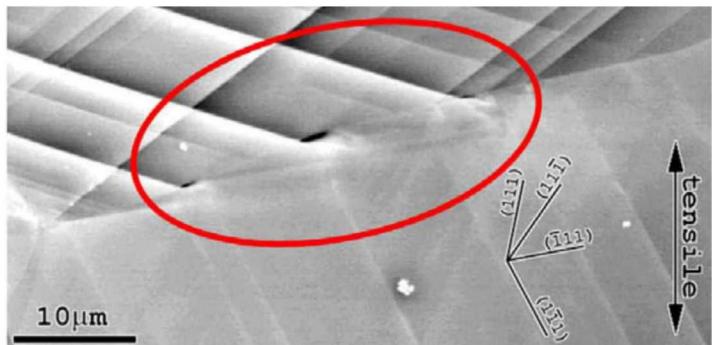


Wilsdorf, HGF. *Acta Metallurgica*, 1982

Early studies of the ductile rupture of single crystals demonstrated that voids nucleate at deformation-induced boundaries. Vacancy clusters observed near fracture surface in some materials. **Do dislocation boundaries and, maybe even vacancy clusters, play a role in void nucleation in bulk metals?**

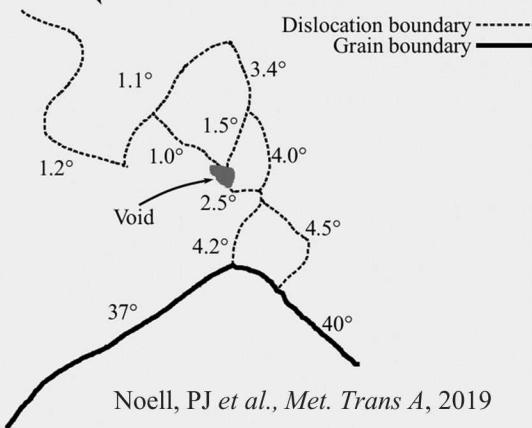
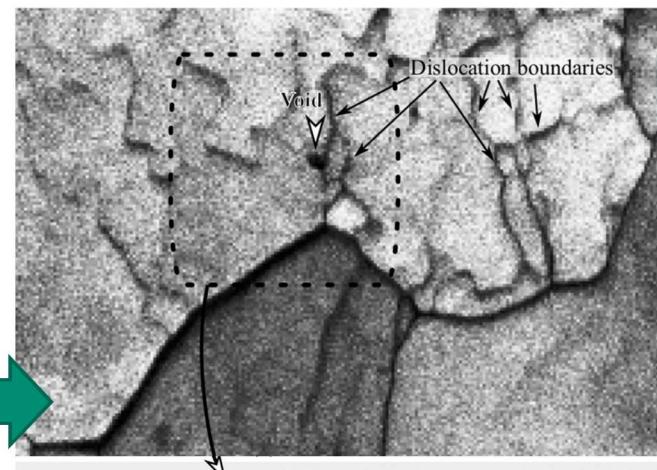
The microstructural features associated with particle-free void nucleation in bulk materials

Microcracks in TiAl correlated with twin intersections.

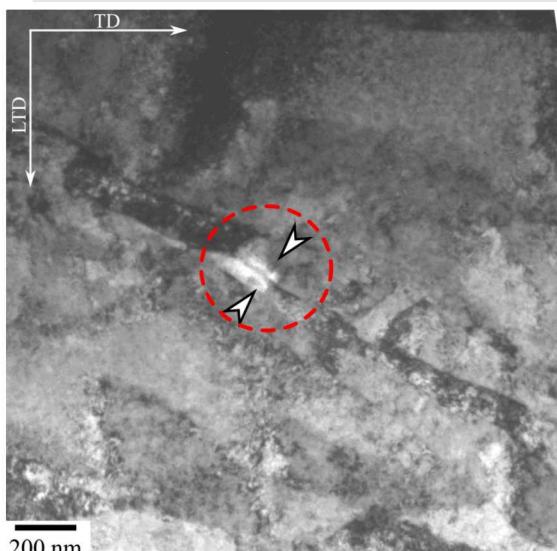


Void nucleation in Al occurred at dislocation cell walls, suppressed by DRX

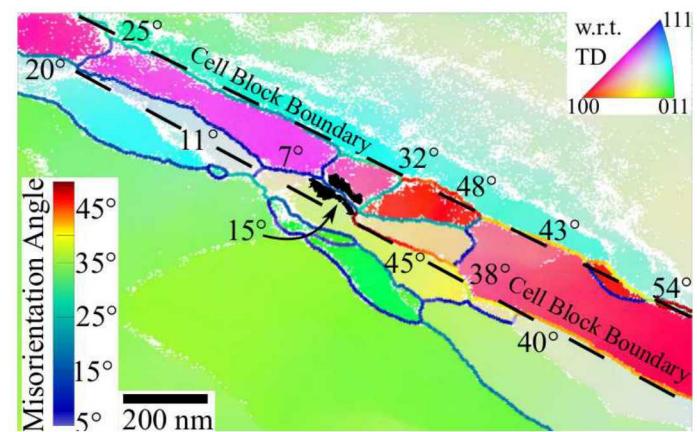
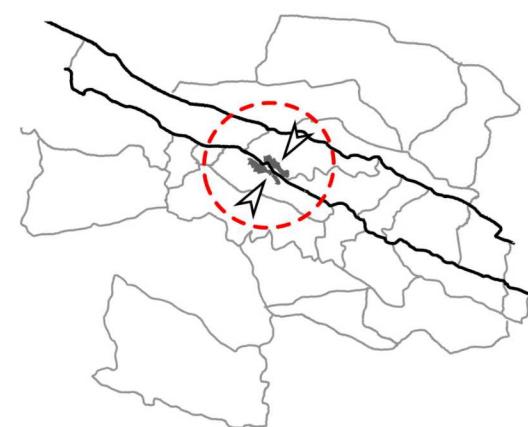
- In pure metals that deform by twinning, voids nucleate at grain-boundary/twin intersections and twin/twin intersections
- In both pure BCC (Ta) and FCC (Al) materials that deform by slip, voids nucleated at dislocation boundaries
- In high-purity Al, void nucleation was suppressed by dynamic recrystallization, which erased dislocation boundaries



Noell, PJ *et al.*, *Met. Trans A*, 2019



Void nucleation in Ta primarily occurred at cell block boundaries

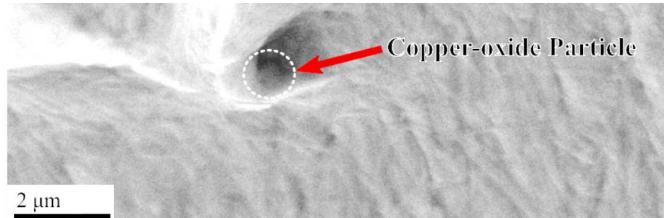


Noell, PJ et al., *Acta Mat.* 2017

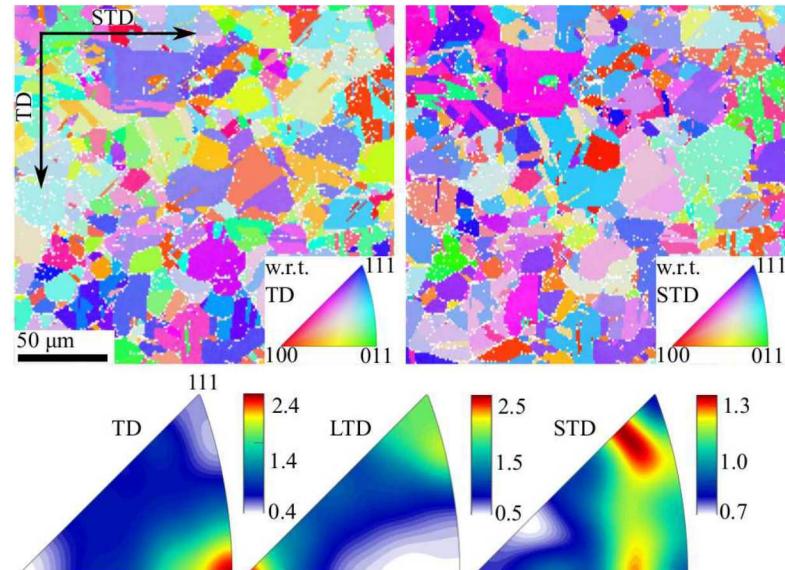
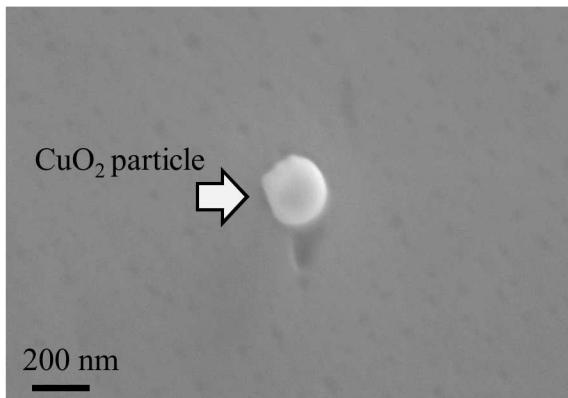
Do deformation-induced defect structures also enable the formation of microscale voids in materials that contain second-phase particles?

Material: 99.9% Cu sheet material, annealed after rolling

Contains \approx 200 to 500 nm copper oxide inclusions, \approx 1 inclusion per $8000 \mu\text{m}^3$ of material



Images of (above) a CuO_2 particle on the fracture surface and (below) a CuO_2 particle within a void



EBSD indicates that the as-received microstructure contained weakly-textured, equiaxed grains (8.7 μm diameter)

1. What deformation-induced defect structures are associated with the formation of microscale voids?
2. Do dislocation boundaries influence the formation of voids at inclusions?
3. Are copper oxide inclusions the primary nucleation site for voids in this material?



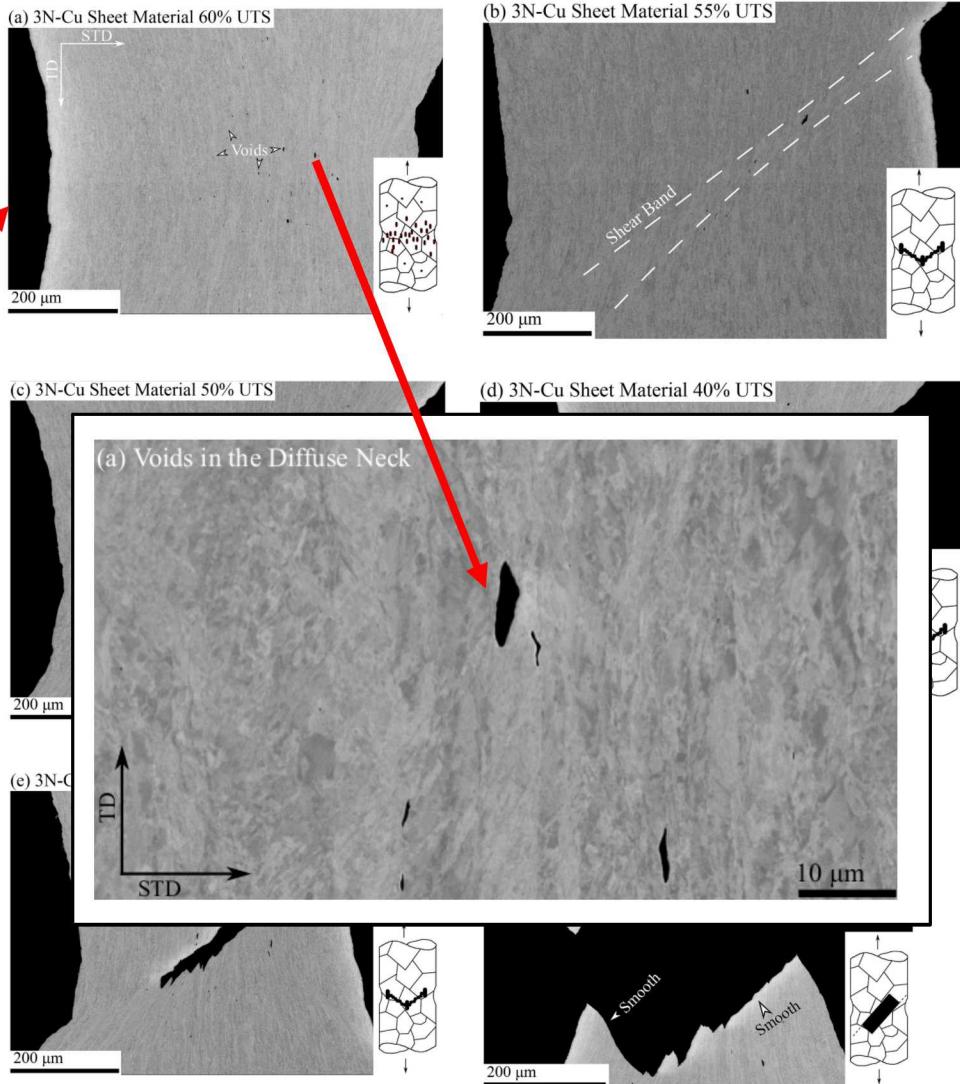
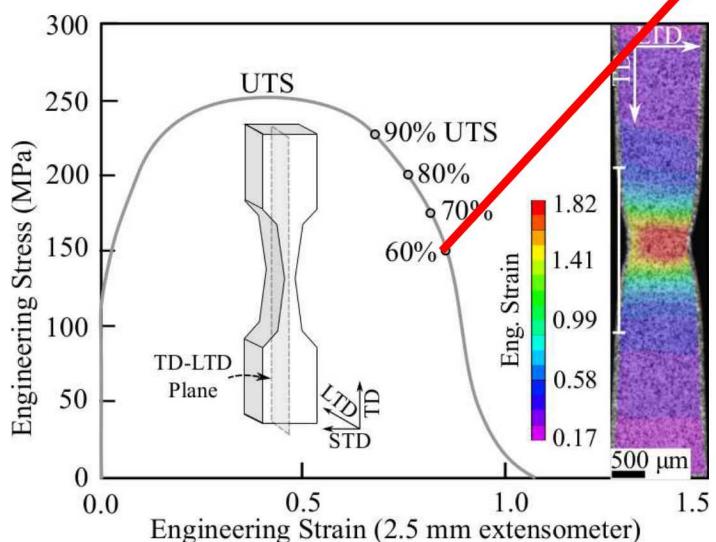
- Failure process
- Defect structures in the deformed material
- The origins of microscale voids



Ductile rupture of Copper

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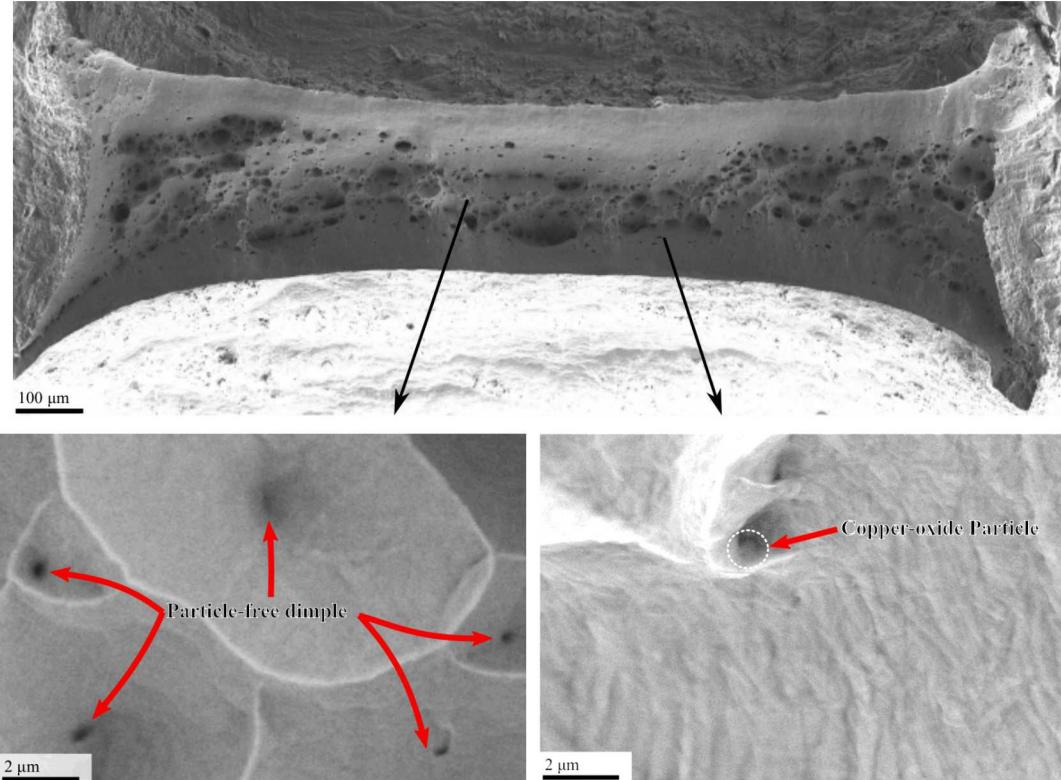
Noell, et al., *Acta Materialia* (2018)



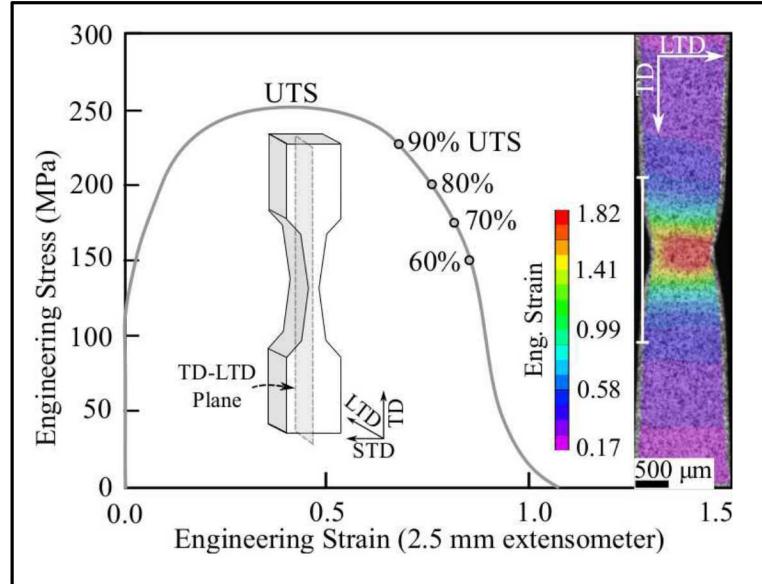
- Previous study using this material revealed that voids nucleate and grow within the diffuse neck of this material, coalesce along a shear band, and create an “Orowan void” that grows until failure
- *But where do the “failure-relevant”, microscale voids come from?*

Producing incipient voids

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Inspection of the fracture surface suggests that most dimples were not associated with copper-oxide inclusions



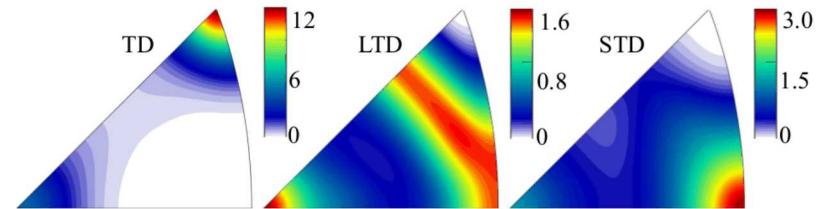
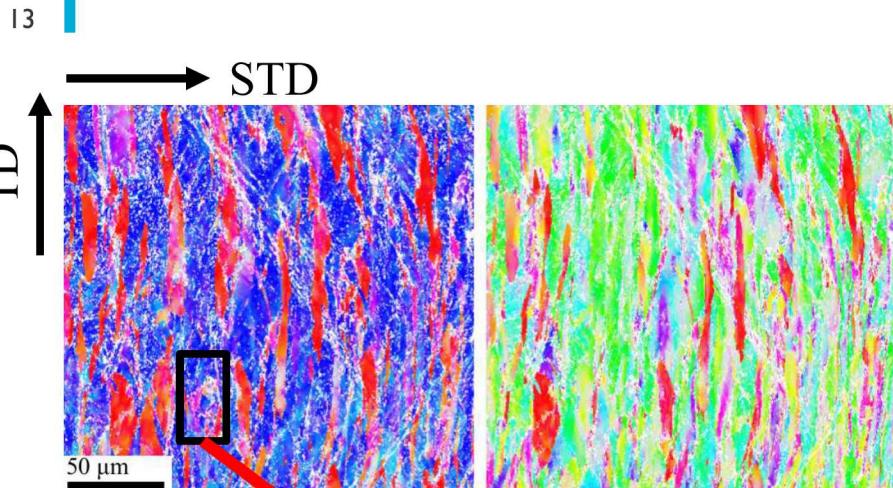
To understand the origins of these voids, interrupted tensile tests used to produce specimens containing incipient voids



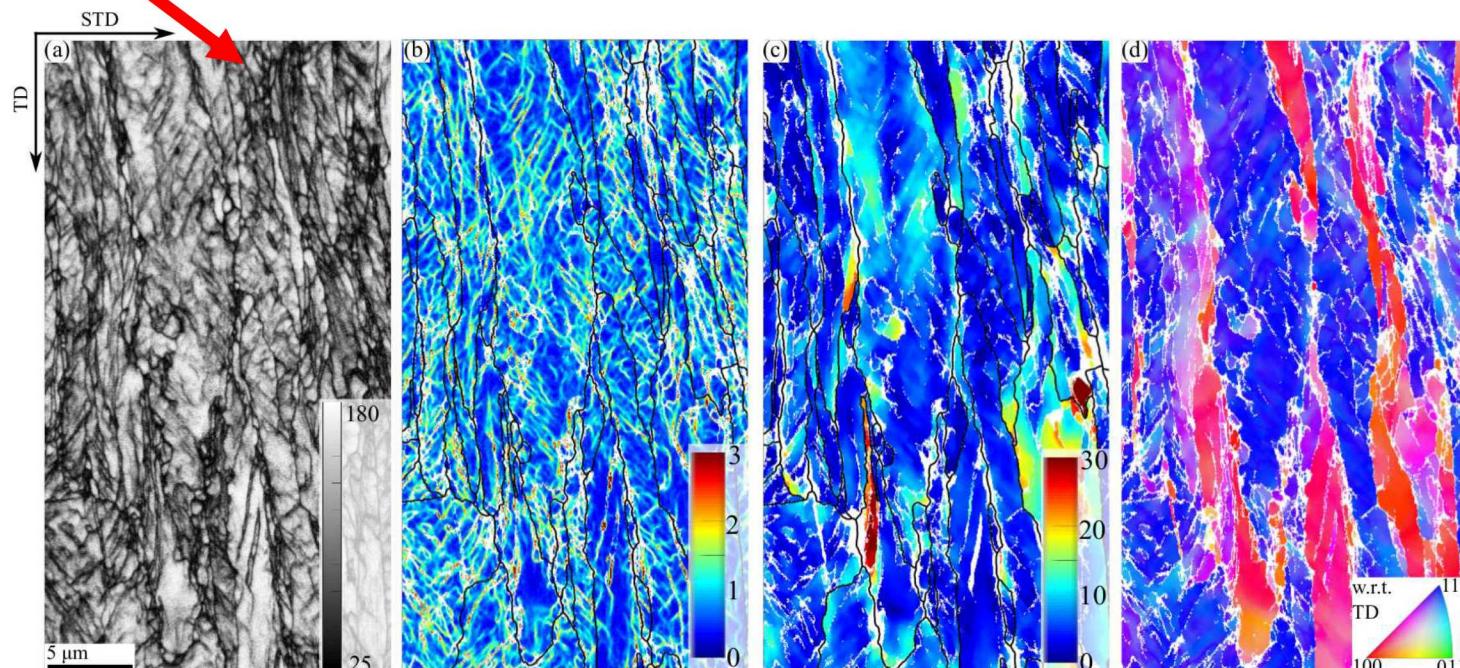
- Failure process
- **Defect structures in the deformed material**
- The origins of microscale voids



The deformed material (60% UTS): deformation boundaries

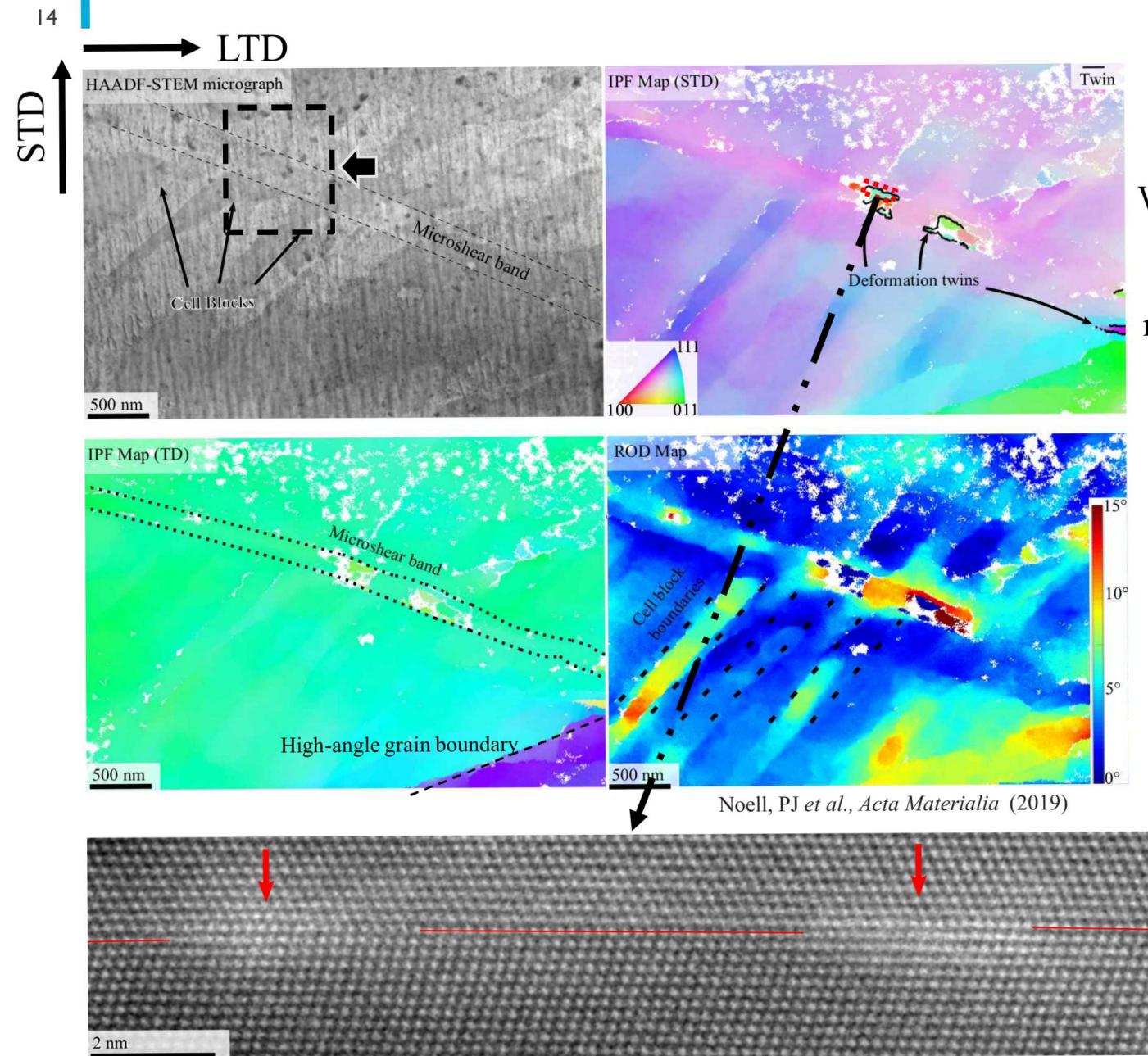


Deformation to 60% of the UTS created elongated grains (along TD), aspect ratio $\approx 4.4:1$. Approximate strain in center of neck $\approx 150\%-180\%$



EBSD data collected with a 50 nm stepsize reveal dislocation boundaries typical of heavily deformed FCC, including both elongated, lamellar boundaries and equiaxed cell structures

Deformation-induced defect structures at the microscale

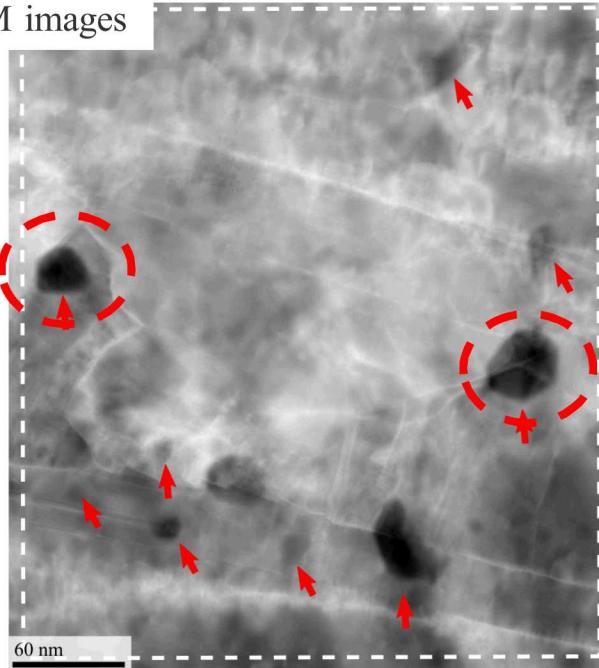
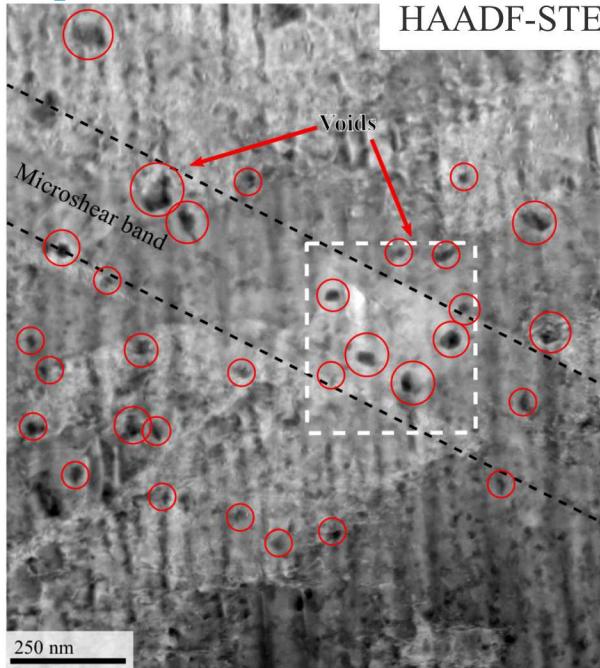


Within the diffuse neck of a 60% UTS specimen, Deformation twins and microshear bands observed at the microscale

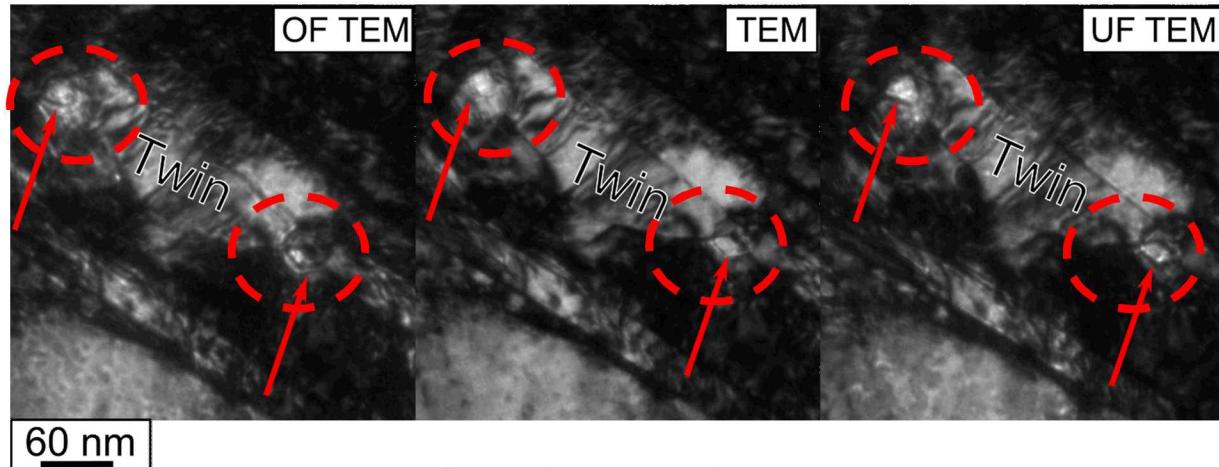
Atomic-resolution detail from HAADF-STEM image of twin boundary shows $(a/6)<112>$ dislocations typical of deformation twins

Vacancy clusters/nanoscale voids

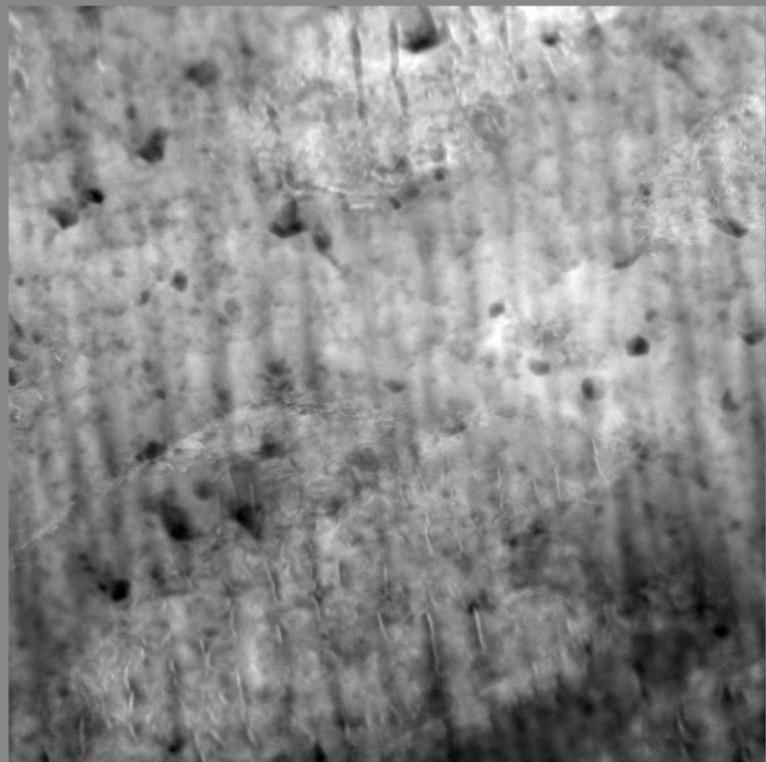
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- Dozens of nanoscale (≈ 10 -100 nm) voids observed both within and outside the microshear band
- Voids are faceted, similar to those created by vacancy condensation following quenching in other materials
- ***No evidence in EDS or diffraction measurements that suggests a change in composition or secondary-phase***



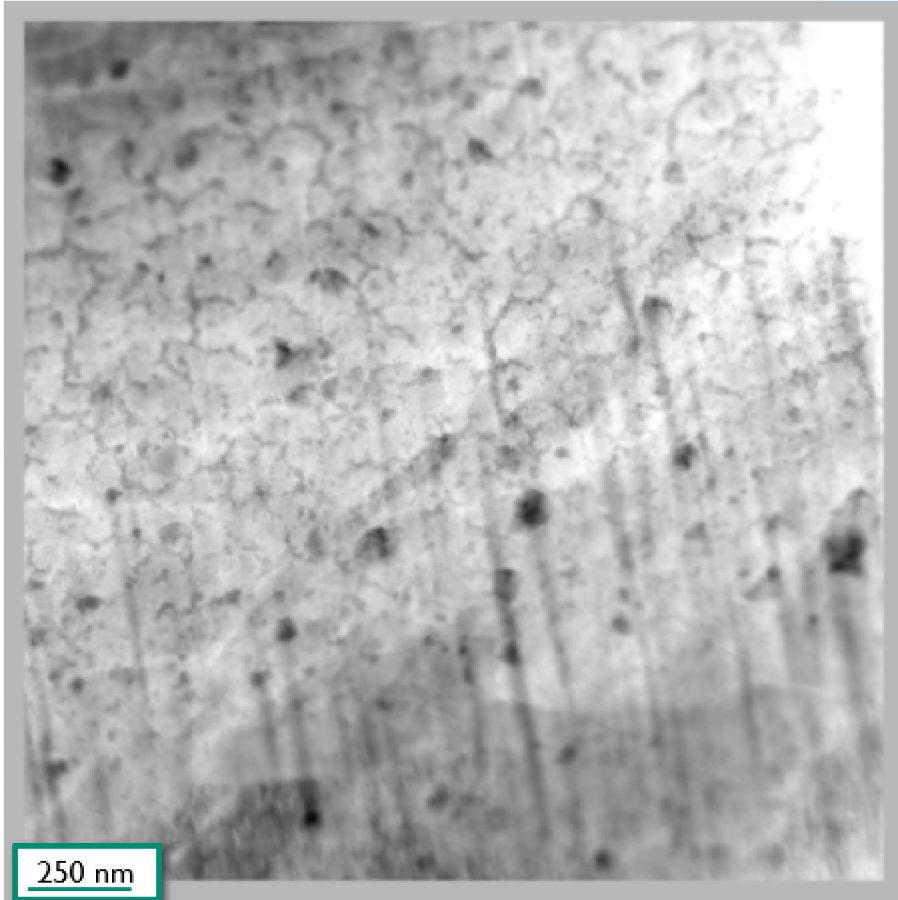
A high density of nanoscale voids appear to be distributed throughout the midplane of the diffuse neck



250 nm

A HAADF-STEM tomographic tilt series reveals mass-thickness contrast (indicative of voids) throughout the sample thickness.

- Estimate from these micrographs suggests a density of 10^2 per μm^2 nanoscale voids
- Observed both within microshear band and outside, suggesting they may be distributed homogenously throughout necked region



250 nm

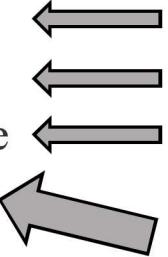
Mass-thickness contrast, indicative of voids, also seen in HAADF-STEM of a second TEM foil (also extracted from the midplane, necked gauge region of a 60% UTS sample)

Proposed void nucleation mechanisms include:

- Particle decohesion
- Particle delamination
- Grain boundary cleavage
- Vacancy condensation

Void nuclei will be distributed heterogeneously at specific microstructural features

Void nuclei will be fairly homogeneous



- Experimental measurements of Cu at $\epsilon = 60\%$ between 10^{-7} and 10^{-4} (T. Ungar, *et al.*, *Mat Sci Eng A*, (2007))
- Vacancy concentration in neck at $\epsilon \approx 150\%$ predicted to be $\approx 10^{-4}$

$$c_v = \frac{A}{G} \int_0^{\epsilon} \sigma d\epsilon$$

C_v - vacancy concentration
A - constant (A~0.1)
G - shear modulus

- Void nucleation in this material is **not** controlled by inclusions or second-phase particles
- This result strongly suggests that vacancy condensation contributes to void nucleation

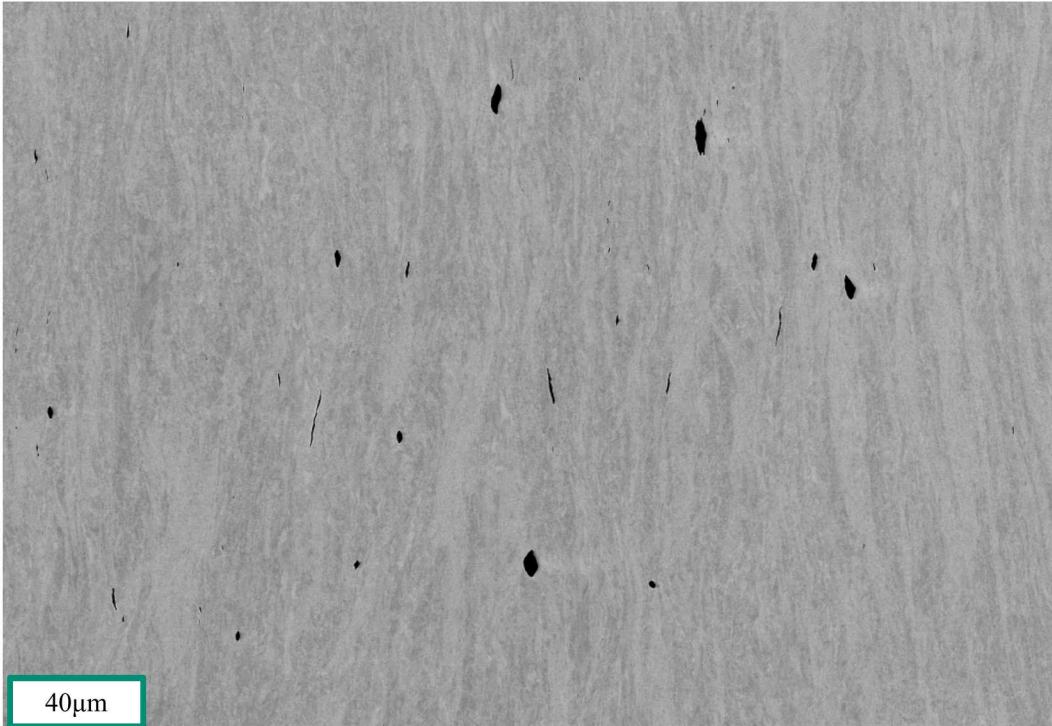
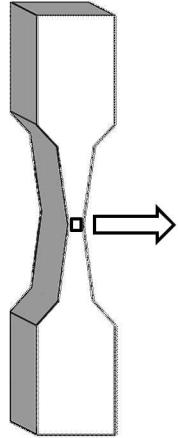


- ❑ Failure process
- ❑ Defect structures in the deformed material
- ❑ The origins of microscale voids



What happens to the nanoscale voids?

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Area density of microscale voids in the center of the neck of a 60% UTS sample was ≈ 1 per $400 \mu\text{m}^2$. Conservatively, only 1 in 10,000 nanoscale voids grows to the microscale. *Void nuclei thus appear to be overseeded!*

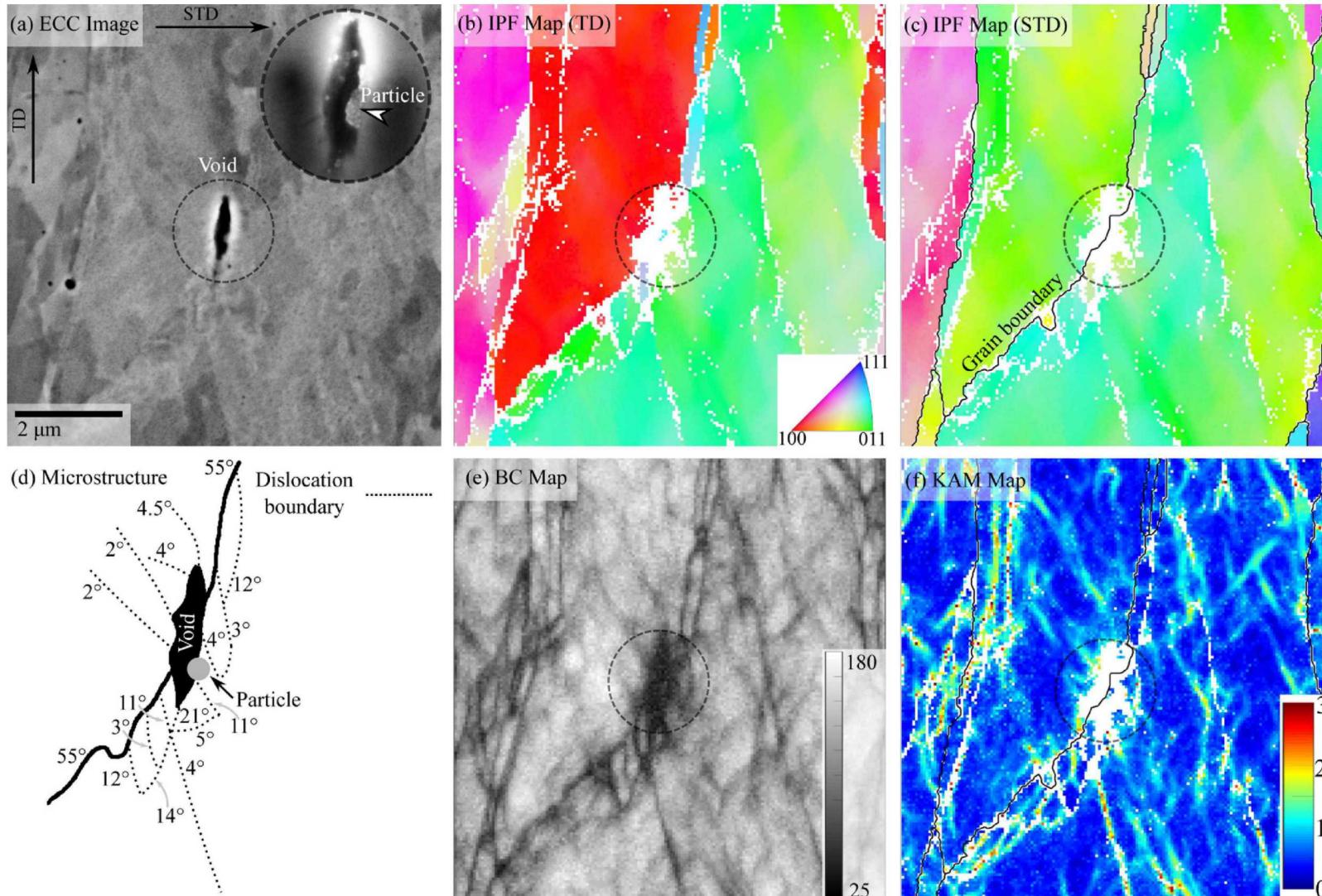
Why do some voids grow while the vast majority stagnate?

22 incipient (< 2 μm) voids characterized using a combination of EBSD (20) and TEM (2)

- All voids associated with a deformation-induced dislocation boundary
- 13 microscale voids were particle free and emerged within a grain
- 6 microscale voids emerged at the inclusion-free intersection between a grain boundary and one or more dislocation boundaries
- 5 microscale voids emerged at an inclusion that was intersected by a dislocation boundary and, in most cases, a grain boundary

Particle-stimulated voids: microscale

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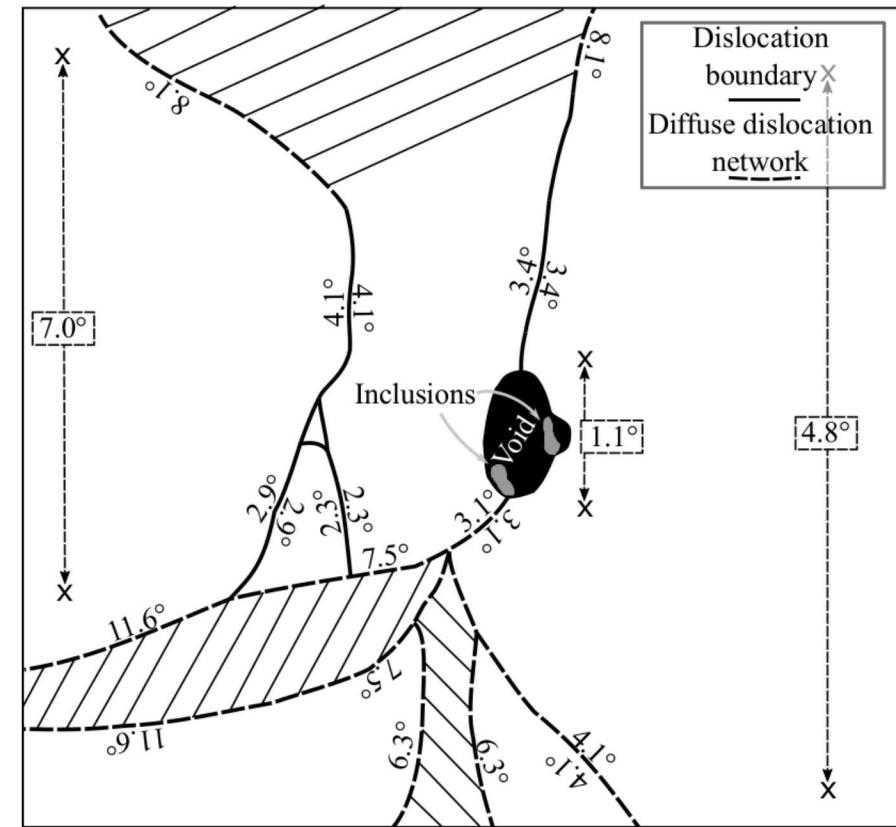
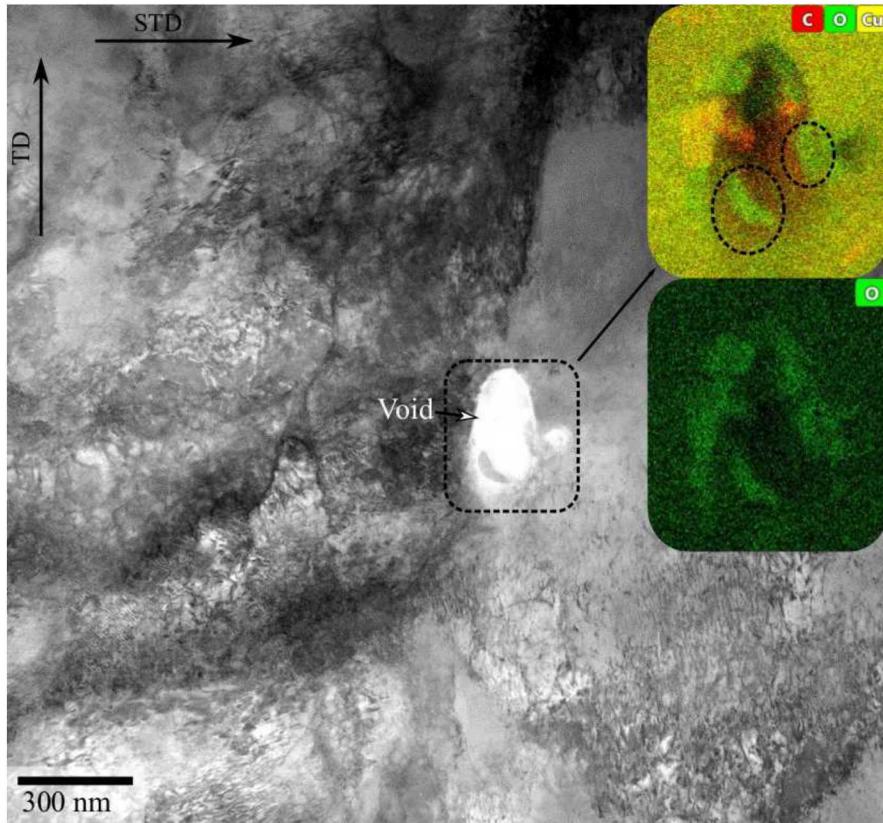


Typical void that nucleated at a particle is shown

- Intersected by multiple cell block boundaries

Particle-stimulated voids: nanoscale

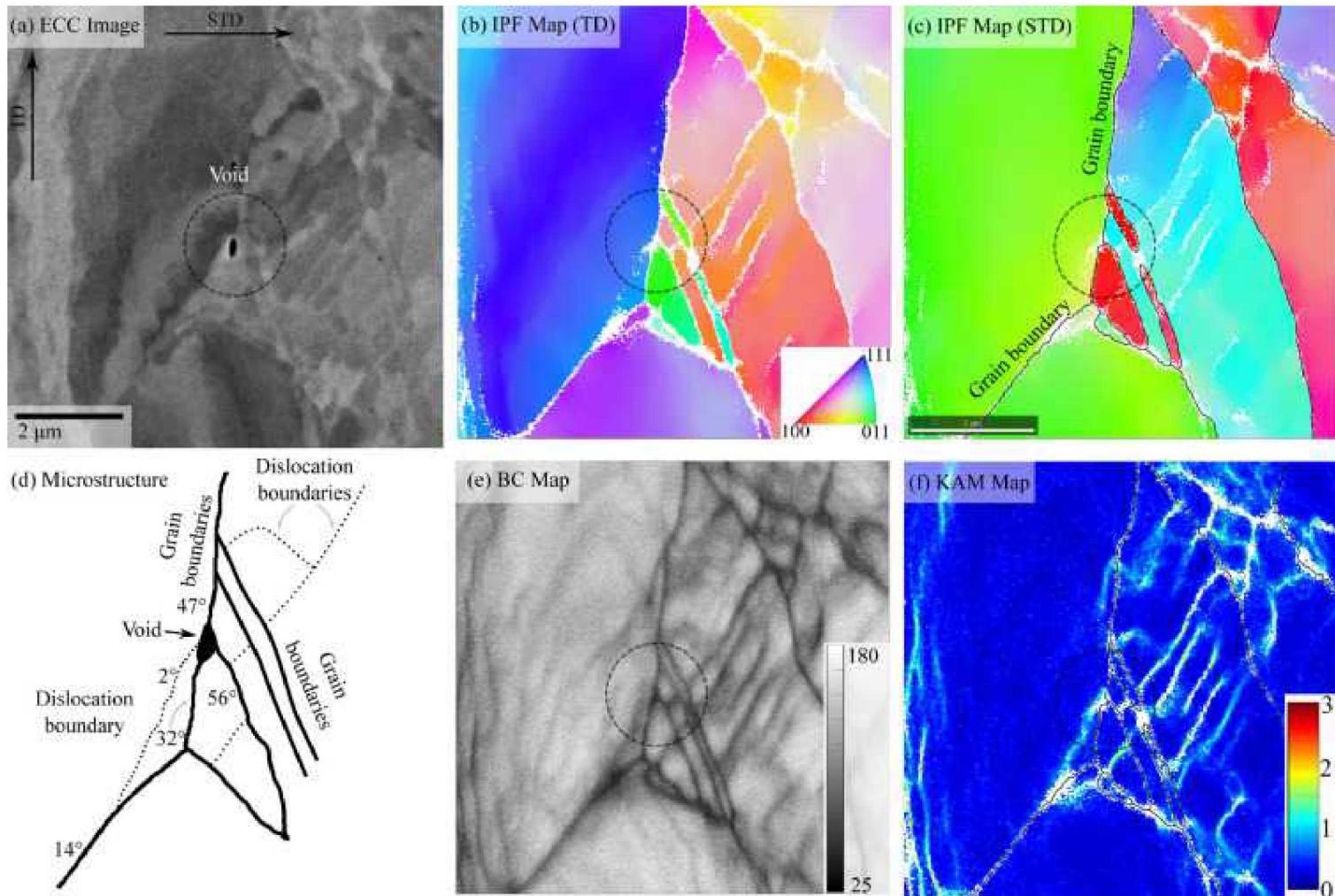
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Even at the nanoscale, it is clear that dislocation boundaries are associated with the emergence of voids that are associated with particles

Grain-boundary, inclusion-free voids

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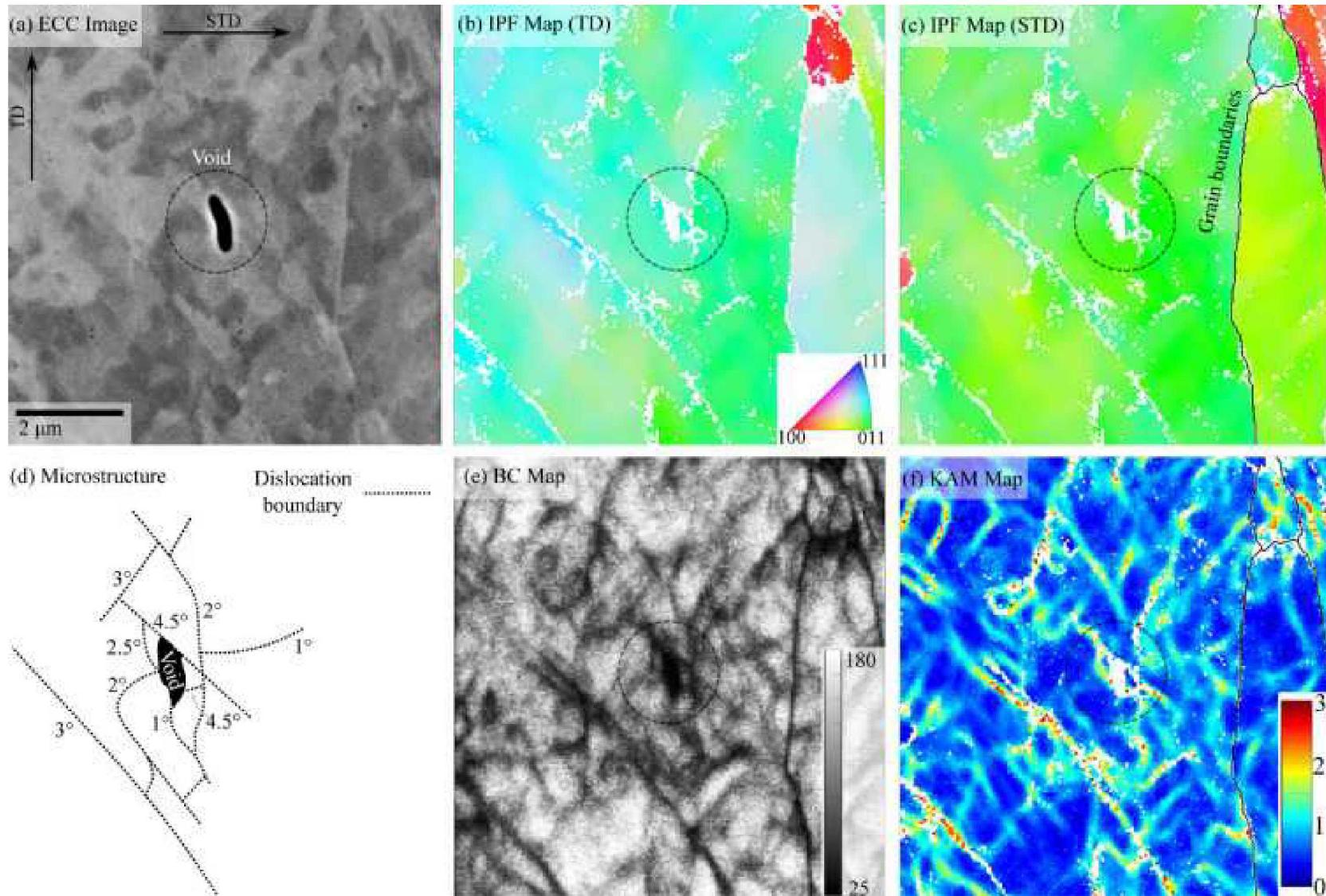


Typical void that nucleated at a grain boundary is shown

- Void emerged at the intersection between a grain boundary triple point and a cell block boundary

Intragranular, inclusion-free voids

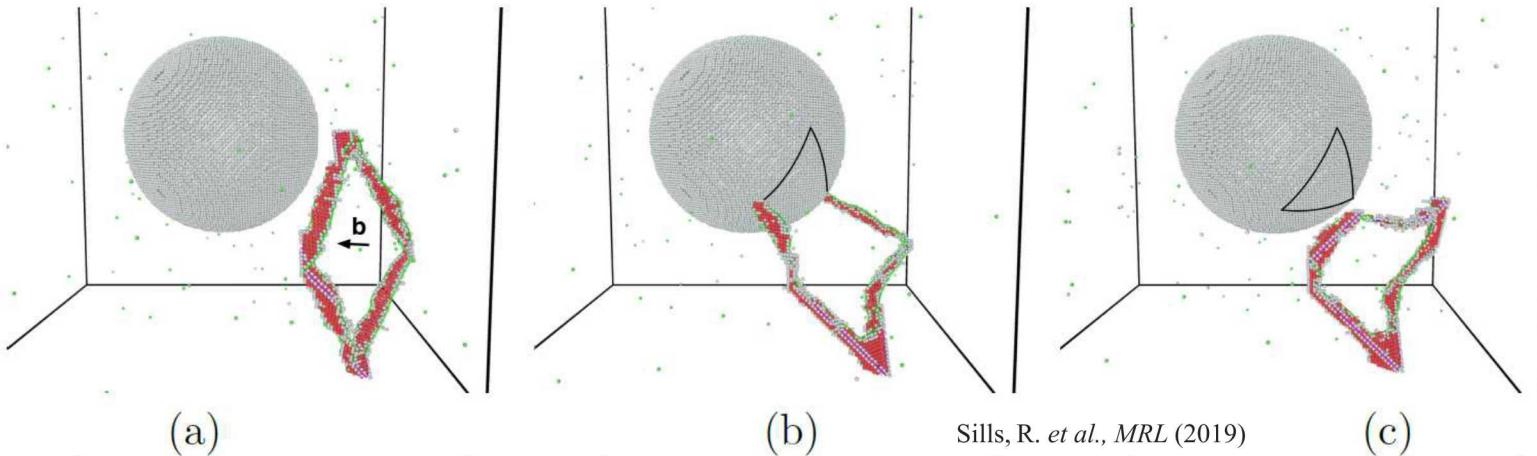
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Most microscale voids emerged within grains and were not associated with inclusions or particles

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These observations suggest void emergence to the microscale depends on dislocation structures



Sills, R. et al., MRL (2019)

- MD simulations of void growth by dislocation absorption suggest that this is a realistic mechanism for the growth of nanoscale voids
- This mechanism is only realistic if the nanovoid is surrounded by a high-density of mobile dislocations, which is expected if the nanovoid is intersected by one or more dislocation boundaries

Conclusions

1. At what microstructural features do microscale voids originate? *Are inclusions the primary nucleation site?*
2. What deformation-induced defect structures are associated with the formation of microscale voids at inclusions?

- Voids nucleated at the nanoscale, perhaps by vacancy condensation, and were distributed fairly homogeneously within the midplane of the necked gauge region
- *Void nuclei appear to be overseeded.* The formation of microscale voids thus controlled by the emergence of nanoscale voids to the microscale (early stages of void growth) and *not* by void nucleation.
- The early stages of growth depend primarily on the location of the void nuclei and appears to be controlled by the dislocation structures that intersect the void.
- *The evolution of dislocation structures appears to control the formation of failure-critical voids.*

Are “vacancy clusters” just a specimen preparation artifact?

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- TEM foils extracted from fillet between gauge and grip



(a)
away from the Ga melt surface layer.

(b)

(c)



This observation suggests that any nanoscale defect content smaller than ≈ 3 nm may be an artifact of specimen preparation, but that ***the >10 nm voids observed in the deformed microstructure are real, intrinsic features of the microstructure***

Extremely little mass-thickness contrast observed in tomographic series. Some surface damage is visible at the extreme tilts (central feature due to electron beam depositing material).