

The Peridynamic Model and Nonlocality

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Symposium on Nonlocal Methods and Length Scale Effects

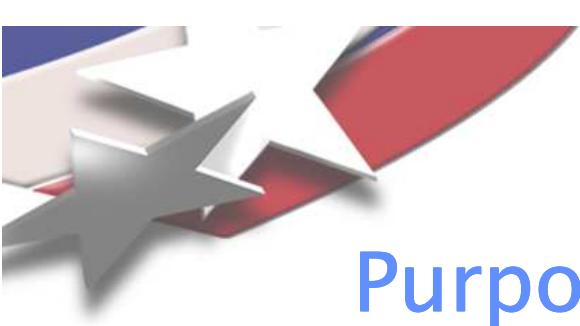
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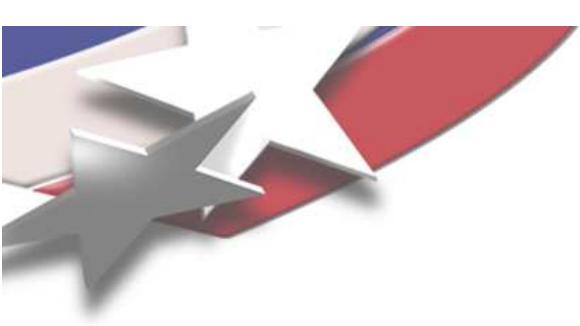




Purposes of the peridynamic model

- To treat material with cracks using the same equations as without cracks.
- To treat discrete particles using the same equations as continua.

- Why do this?
 - The standard theory is not a good tool for modeling cracks.
 - PDEs do not apply on discontinuities or to discrete particles.
 - This leads to the need for special techniques when cracks are present.
 - No natural way to couple atoms to continua.



Strategy

Replace the standard PDEs with integral equations.

- The integral equations involve interaction between points separated by finite distances (nonlocality).
- The integral equations are not derivable from the PDEs.
 - But they converge to the PDEs in the limit of small length scales.

Bond-based peridynamic model

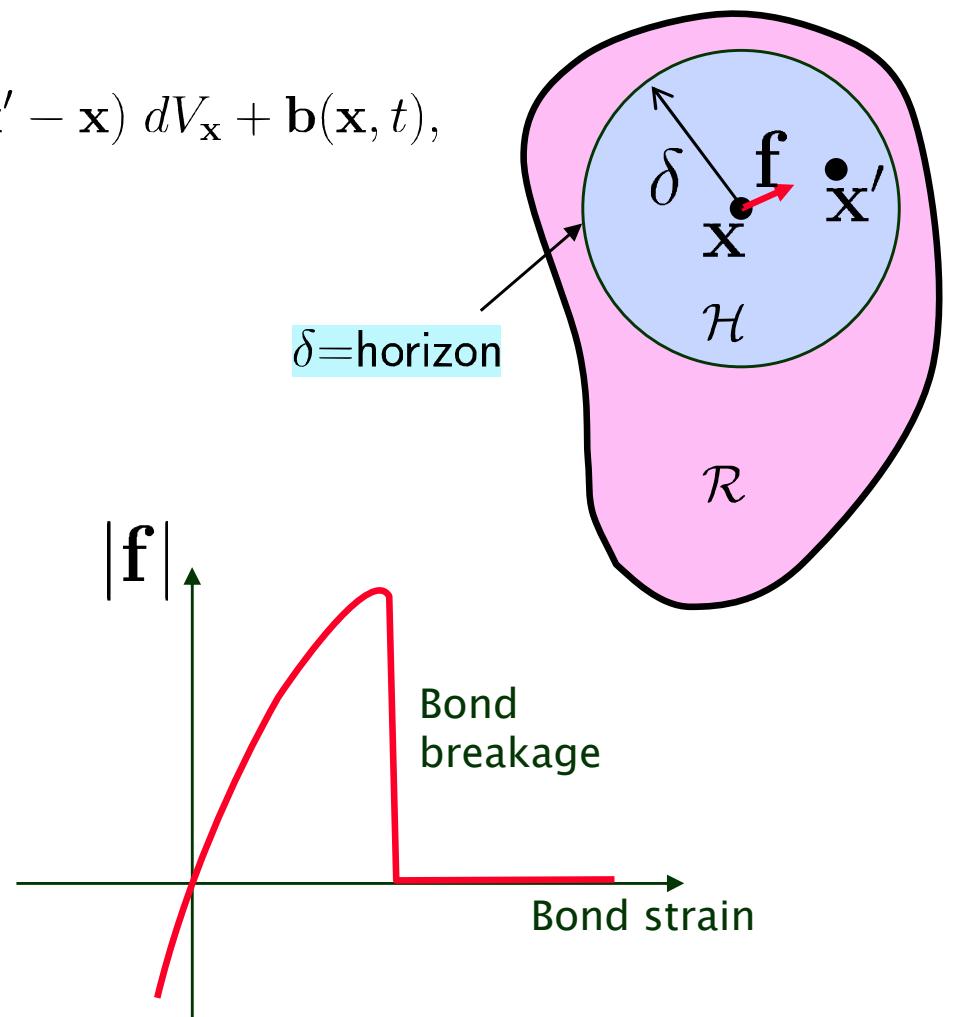
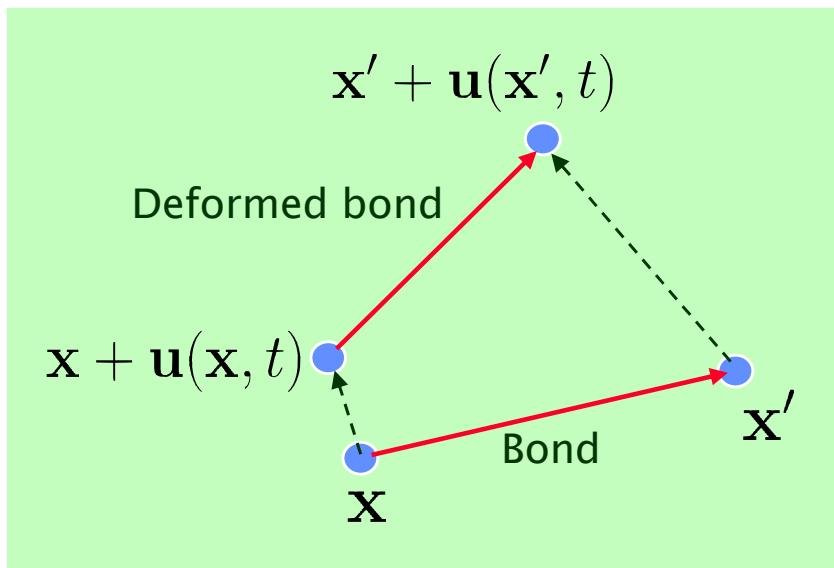
The original (2000) peridynamic model...

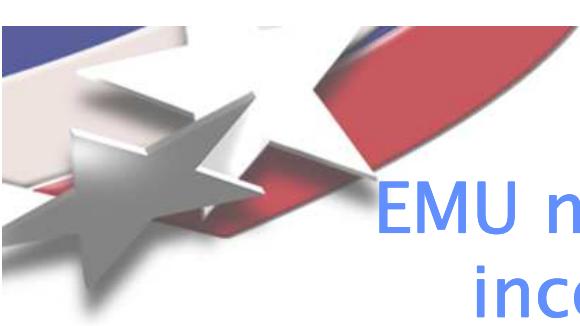
$$\rho \ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{u}(\mathbf{x}', t) - \mathbf{u}(\mathbf{x}, t), \mathbf{x}' - \mathbf{x}) dV_{\mathbf{x}} + \mathbf{b}(\mathbf{x}, t),$$

\mathbf{u} =displacement, \mathbf{b} =body force density

\mathbf{f} =pairwise force function (force/volume²)

Sums up the forces that all the \mathbf{x}' exert on \mathbf{x} .

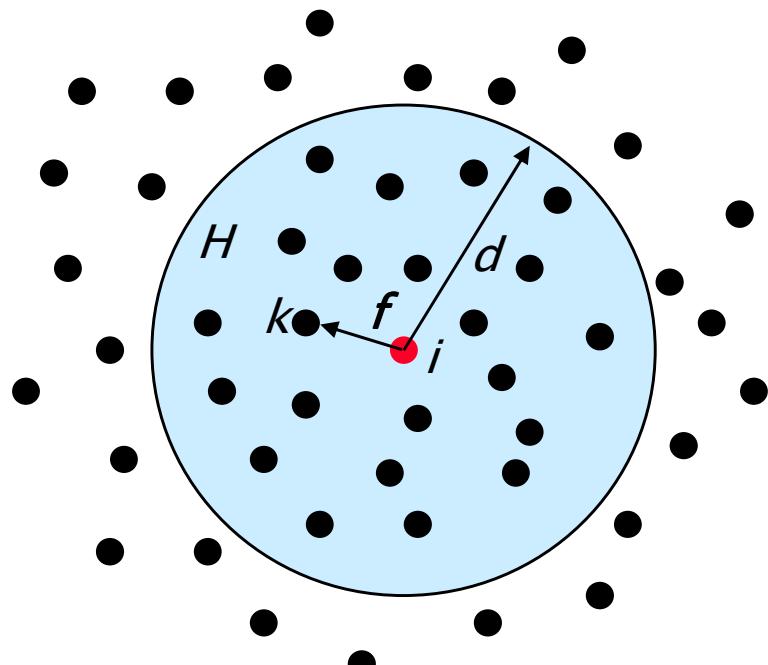




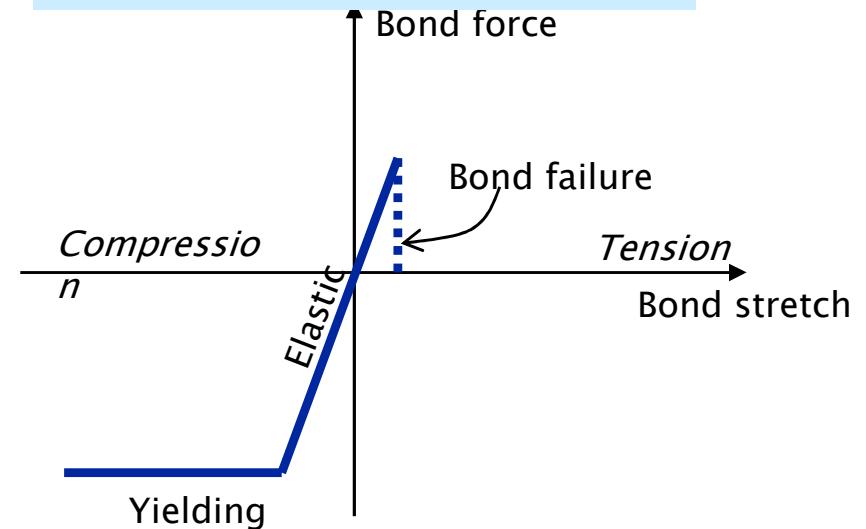
EMU numerical method and material model incorporate damage at the bond level

- Integral is replaced by a finite sum: resulting method is meshless and Lagrangian.
- Parameters come from measurable elastic–plastic and fracture data for materials.

$$\rho \ddot{\mathbf{u}}_i^n = \sum_{k \in H} f(\mathbf{u}_k^n - \mathbf{u}_i^n, \mathbf{x}_k - \mathbf{x}_i) \Delta V_i + \mathbf{b}(\mathbf{x}_i, t)$$



All material-specific behavior is contained in the function f .

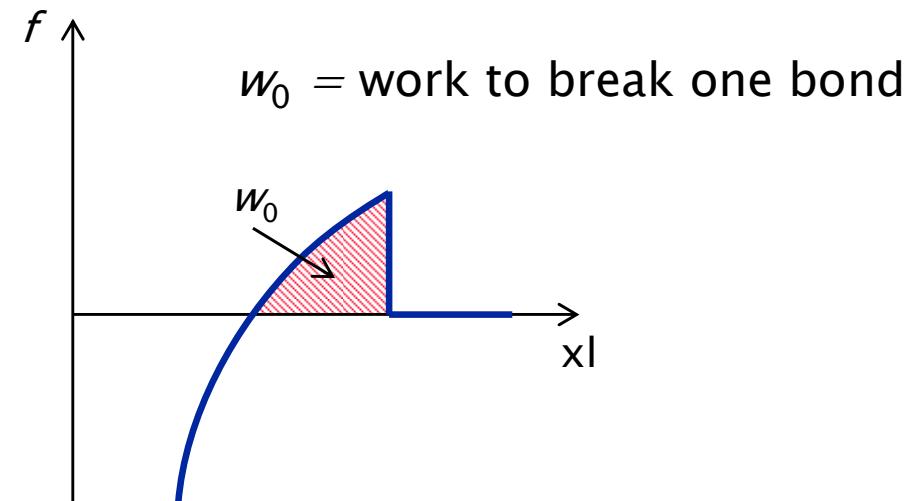
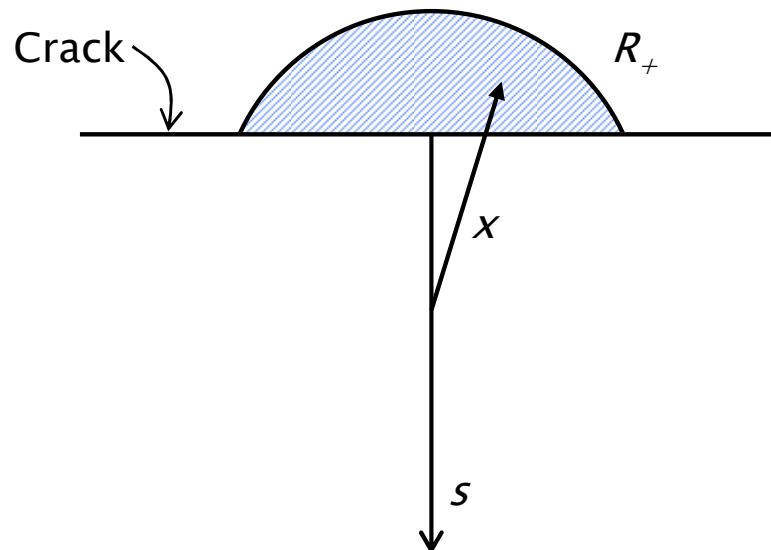




Peridynamic theory: Energy required to advance a crack

- Adding up the work needed to break all bonds across a line yields the energy release rate:

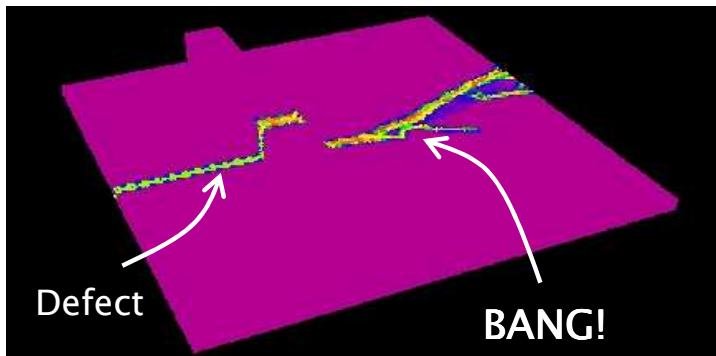
$$G = 2h \int_0^\delta \int_{R_+} w_0 dV ds$$



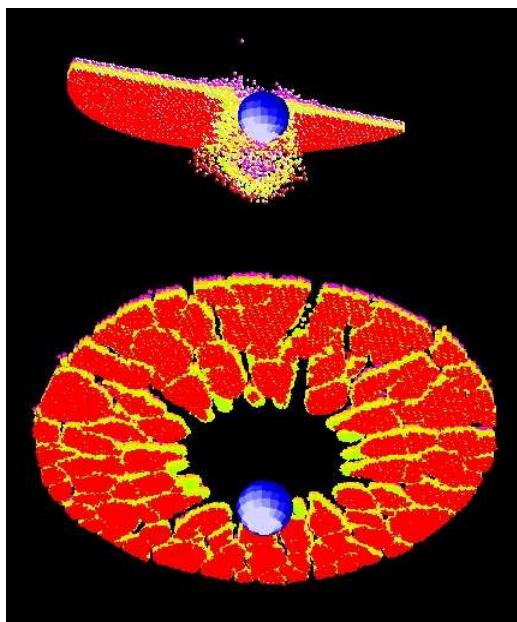
There is also a version of the J-integral that applies in this theory.



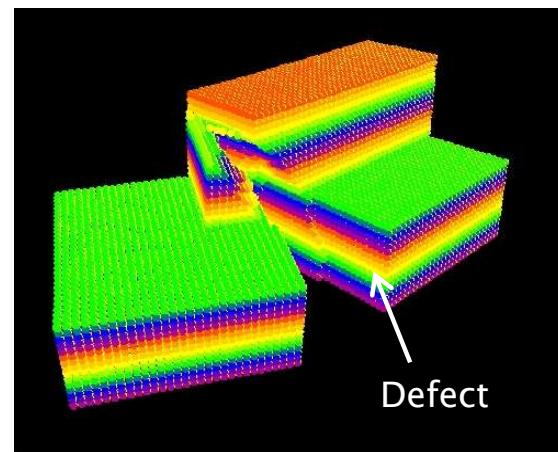
Bond based PD: Fracture and fragmentation



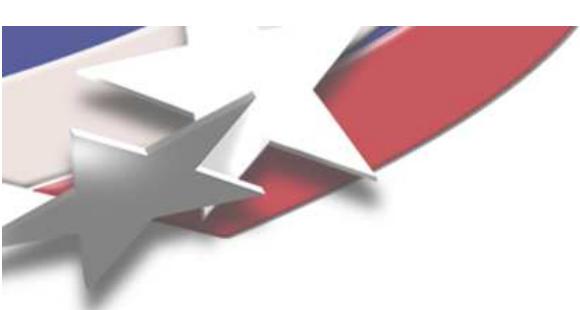
Transition to unstable crack growth



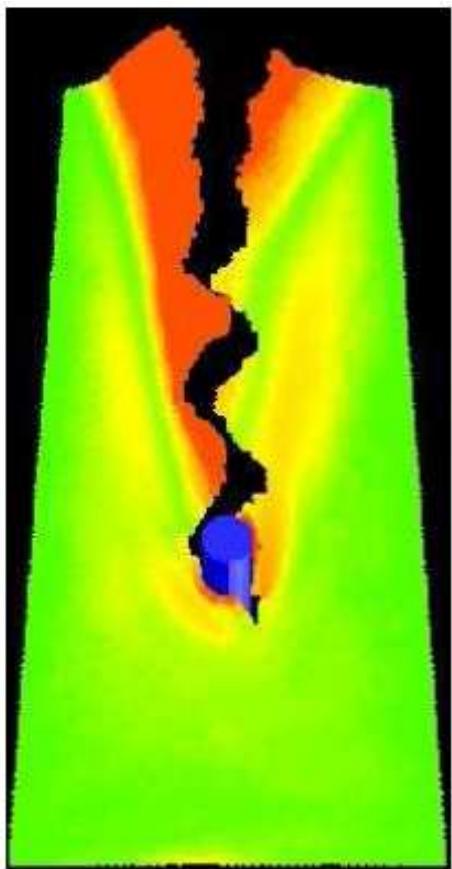
Impact and fragmentation



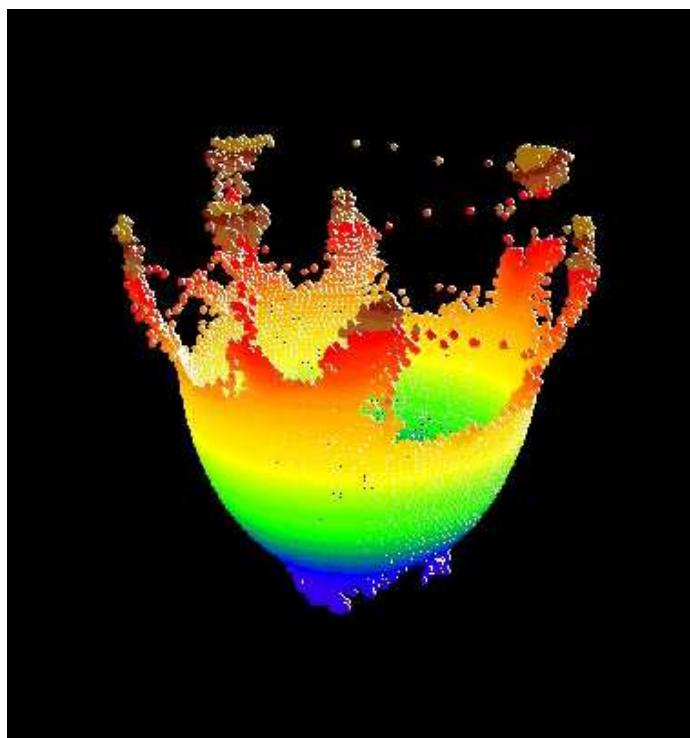
Crack turning in a 3D feature



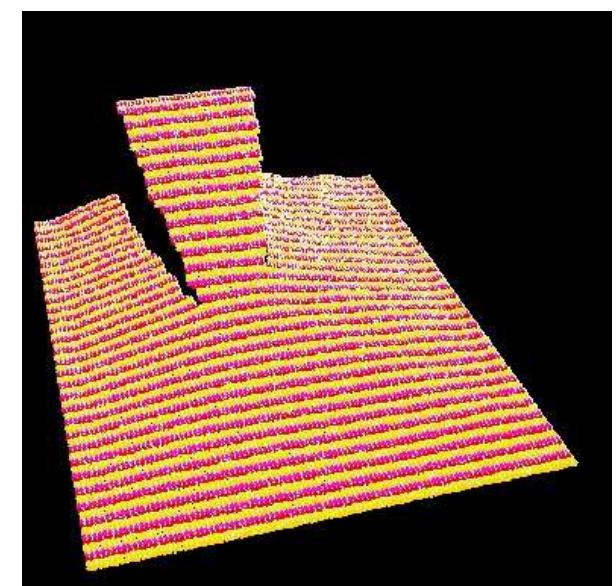
Bond based PD: Elastic membranes



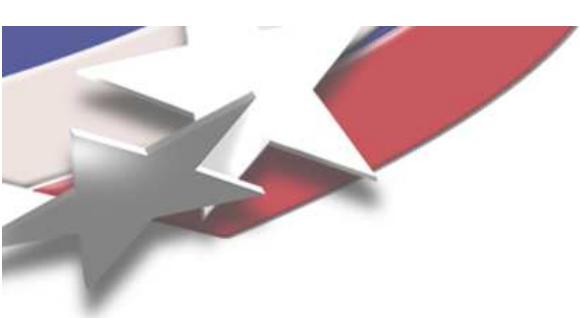
Tearing instability



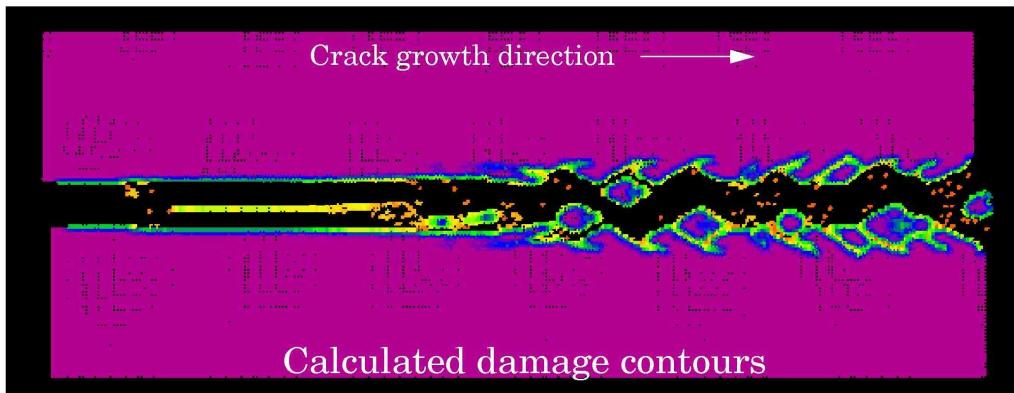
Balloon pop



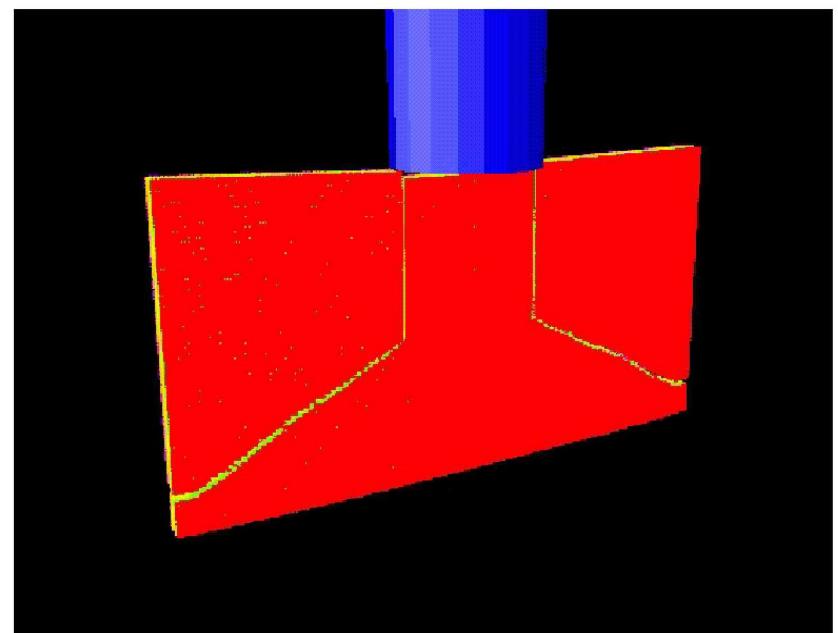
Peeling



Bond based PD: Dynamic fracture



Crack instability in PMMA
(Fineberg & Marder, 1999)

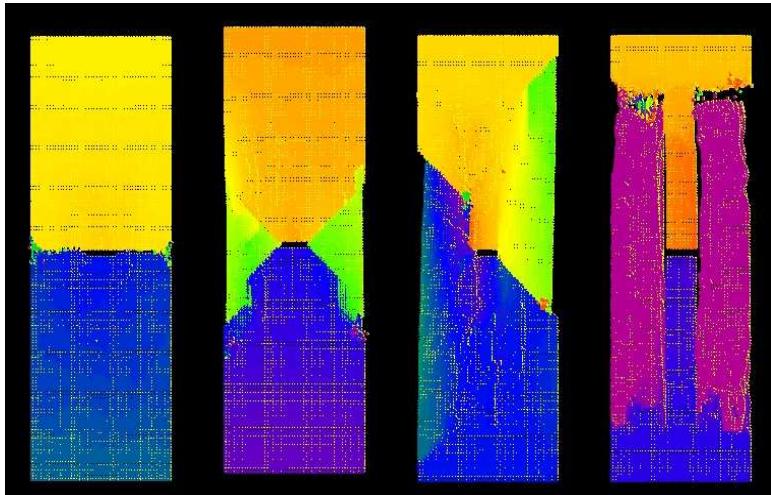


Dynamic fracture in steel
(Kalthoff & Winkler, 1988)

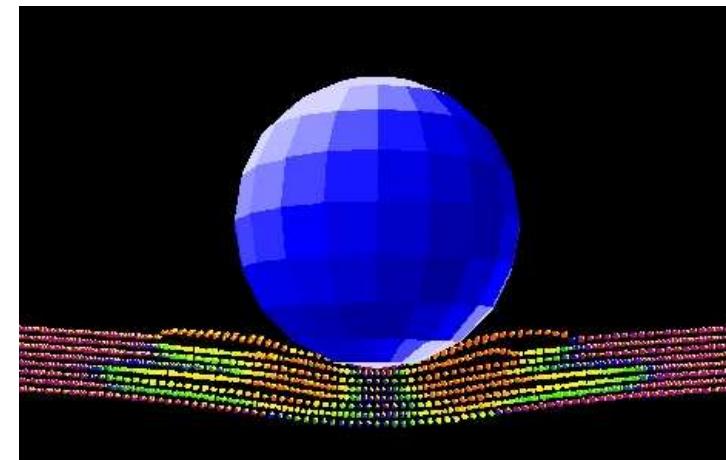


Bond based PD: Damage in composites (Boeing)

- How does the fraction of fibers in each direction affect the direction of crack growth?
- What damage occurs when a composite panel is struck by hail?

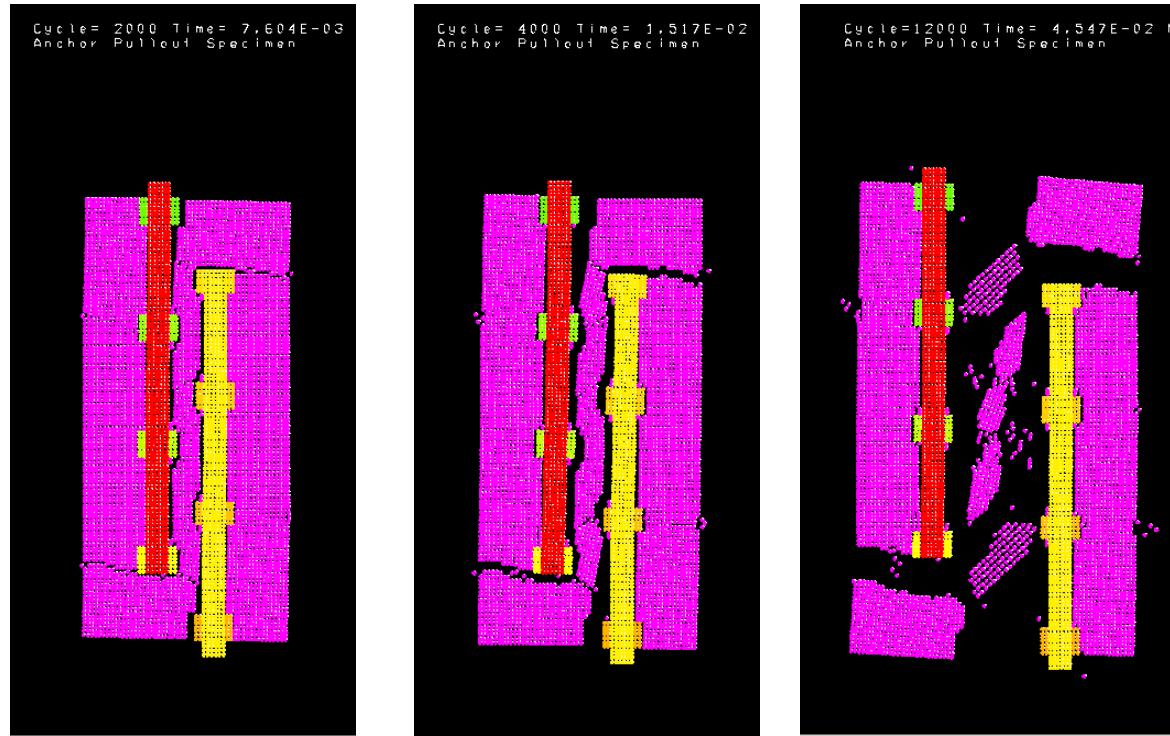


Crack growth in a notched panel

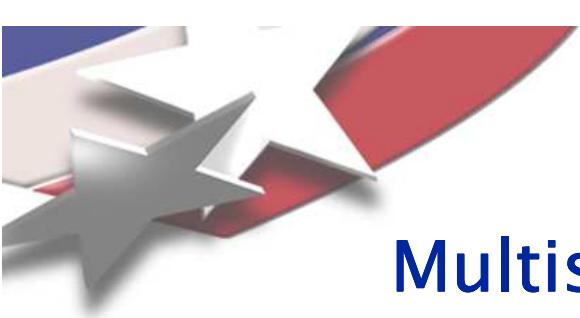


Delamination caused by impact

Splice of ribbed reinforcing bars

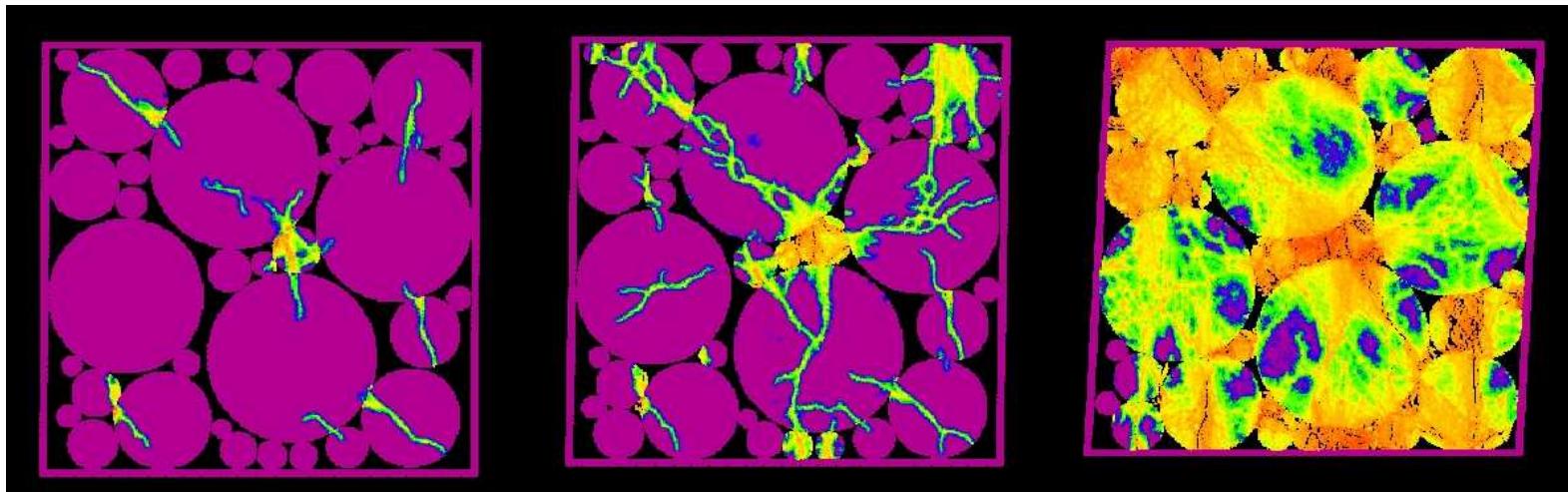


Magnified deformed shapes of splice of reinforcing bars in concrete at three stages
(fine discretization – grid spacing is 0.03 m)



Multiscale high-rate material modeling

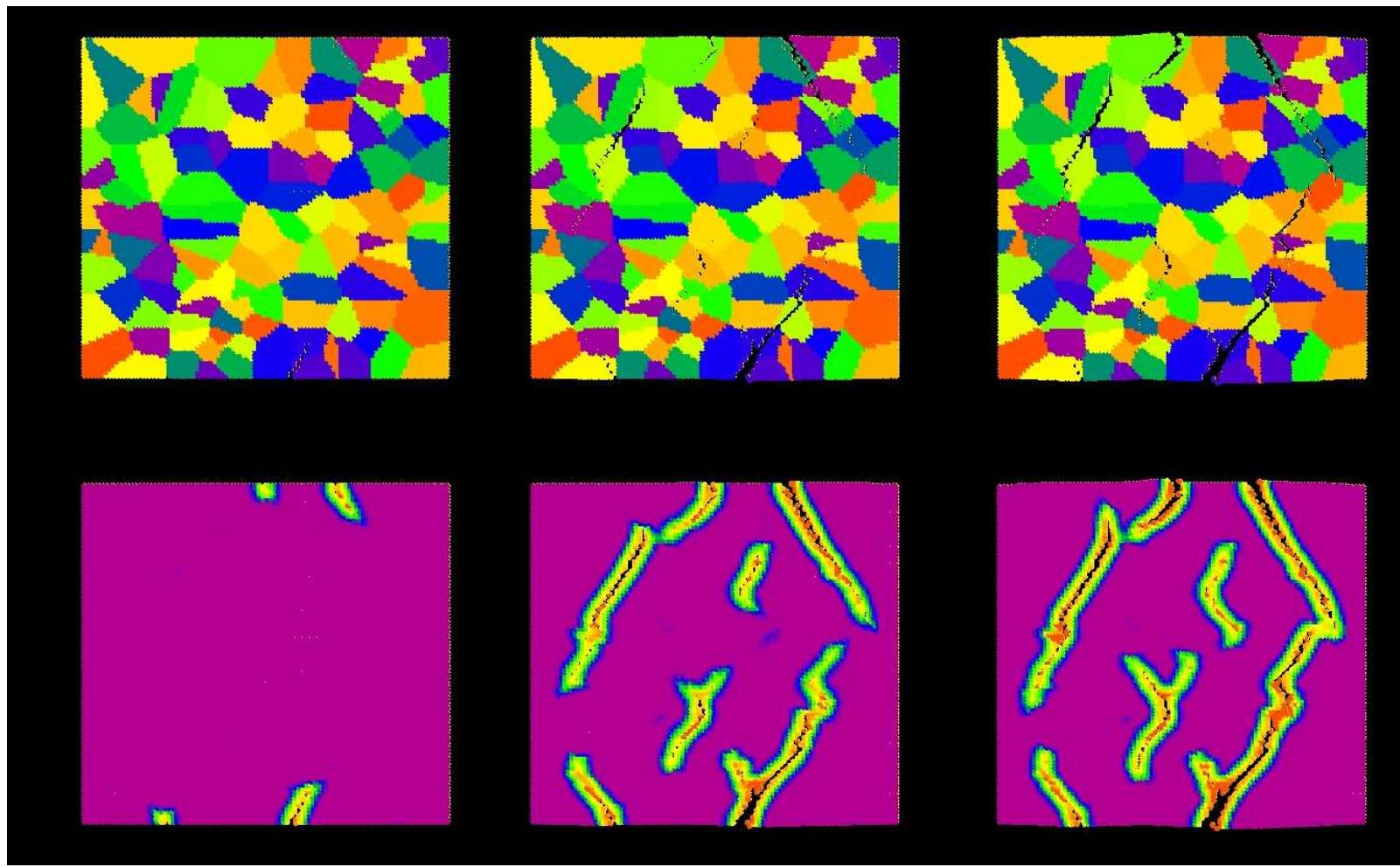
- Grain-scale model includes all relevant physics.
- Statistical treatment leads to macroscale material model.



Combined compression and shear loading at boundaries



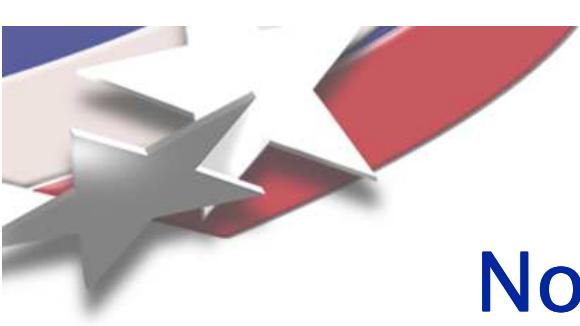
Fracture in a polycrystal



20.5 ms

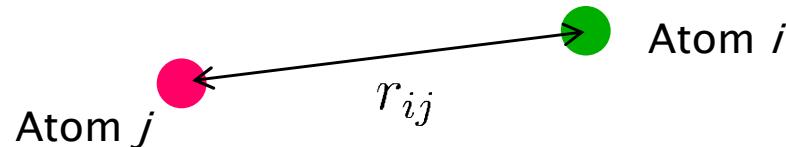
21.3 ms

22.1 ms



Nonlocality and length scales

- Many physical problems have some natural length scale.
 - Sometimes the length scale is obvious, e.g.,
 - Interatomic forces
 - Molecular dynamics cannot be done without nonlocality.



$$F_{ij} \sim \left(\frac{a}{r_{ij}} \right)^{12} - \left(\frac{a}{r_{ij}} \right)^6$$



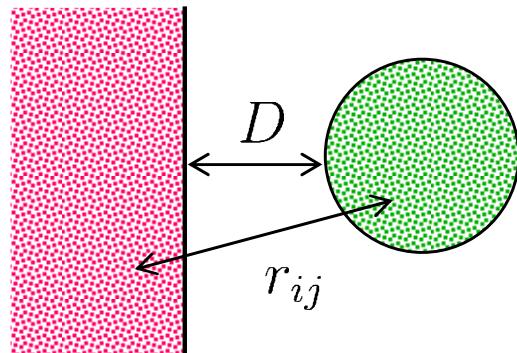
Nonlocality and length scales

- Sometimes the length scale is a little less obvious, e.g.
 - van der Waals forces that lead to longer-range surface forces.
 - Force between a pair of atoms as they are separated:
- Net force between halfspace and a sphere made of many of these atoms* occurs over a much larger length scale:

$$F_{ij} \sim 1/r_{ij}^6$$

- Net force between halfspace and a sphere made of many of these atoms* occurs over a much larger length scale:

$$F_{\text{sphere}} \sim 1/D$$

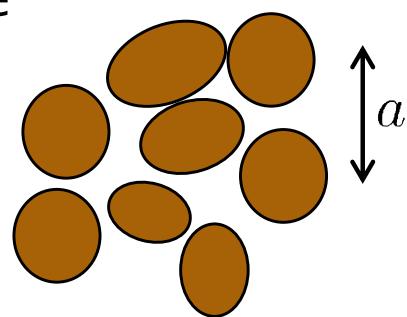


See J. Israelachvili, *Intermolecular and Surfaces Forces*, pp. 177.



Nonlocality and length scales

- Macroscale: sometimes a length scale is determined by heterogeneity, e.g., concrete aggregate size



- Yet damage can occur in a much larger process zone

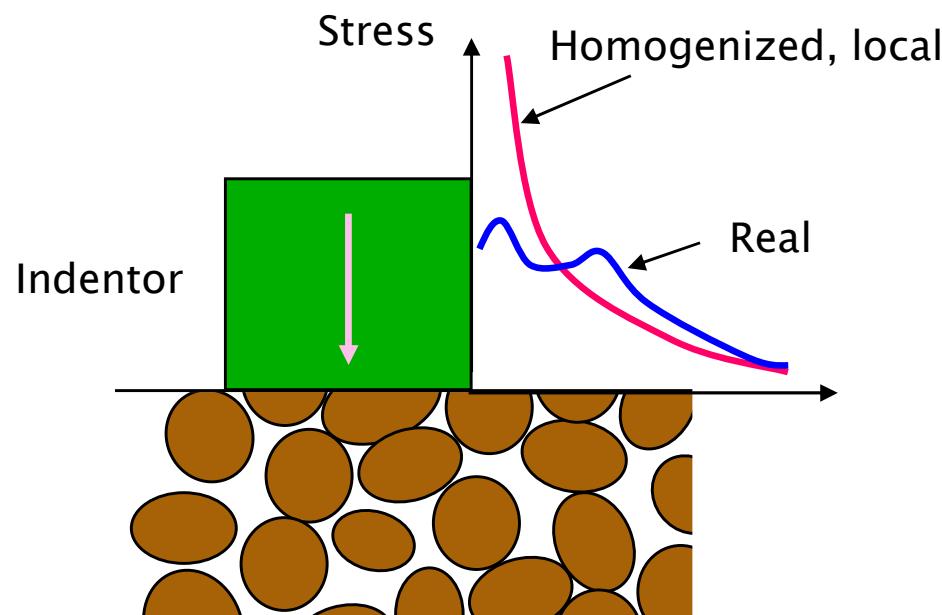


See Bazant,



Nonlocality and length scales

- Homogenization, neglecting the natural length scales of a system, often doesn't give good answers.

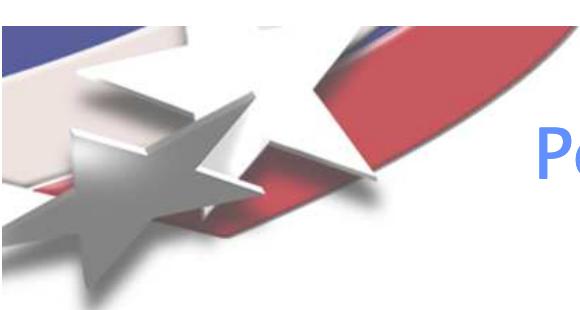




Bond-based peridynamic model: Observations

- Good things:
 - Now have a mathematically consistent way to treat material with cracks.
 - No additional equations are needed to tell cracks what to do.
- Bad things:
 - Independent bonds imply Poisson ratio = 1/4.
 - Can have permanent deformation in bonds, but can't enforce plastic incompressibility.
 - Can't use material models from the standard theory.

Can overcome these limitations by allowing bonds to interact with each other.

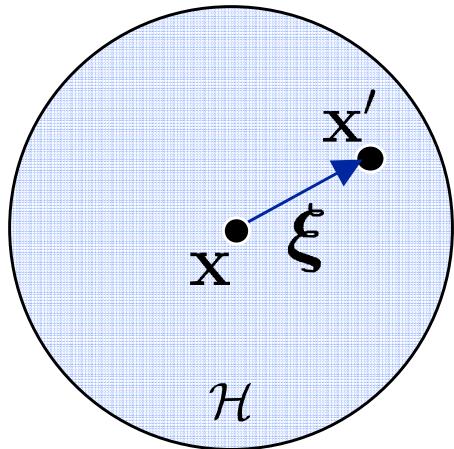


Peridynamic states: Mathematical tool for dealing with collections of bonds

A vector state $\underline{\mathbf{A}}$ is a mapping from \mathcal{H} to \mathbb{R}^3 .

A scalar state \underline{a} is a mapping from \mathcal{H} to \mathbb{R} .

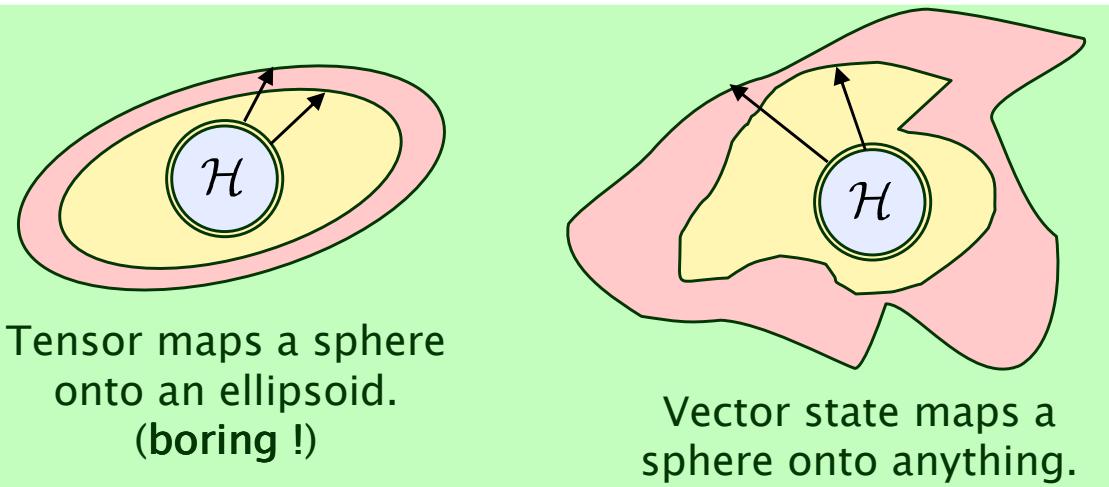
$\xi = \mathbf{x}' - \mathbf{x}$ is a bond. $\underline{\mathbf{A}}(\xi)$ is a vector. $\underline{a}(\xi)$ is a scalar.



\mathcal{H} = set of all bonds connected to \mathbf{x}

A vector state is like a 2nd order tensor, except:

- A vector state can be nonlinear.
- A vector state can be discontinuous.



Tensor maps a sphere
onto an ellipsoid.
(boring !)

Vector state maps a
sphere onto anything.



Dot product of two vector states

Suppose $\underline{\mathbf{A}}$ and $\underline{\mathbf{B}}$ are two vector states. Define the dot product of $\underline{\mathbf{A}}$ and $\underline{\mathbf{B}}$ by

$$\underline{\mathbf{A}} \bullet \underline{\mathbf{B}} = \int_{\mathcal{H}} \underline{\mathbf{A}}\langle \xi \rangle \cdot \underline{\mathbf{B}}\langle \xi \rangle \, dV_{\xi}.$$

The set of all vector states \mathcal{V} is an inner product space (even though each $\underline{\mathbf{A}} \in \mathcal{V}$ is a nonlinear mapping.)



Functions of states and Frechet derivatives

Suppose $\Psi(\cdot)$ is a scalar-valued function of a vector state $\underline{\mathbf{A}}$. For any differential $d\underline{\mathbf{A}}$ let

$$d\Psi = \Psi(\underline{\mathbf{A}} + d\underline{\mathbf{A}}) - \Psi(\underline{\mathbf{A}}).$$

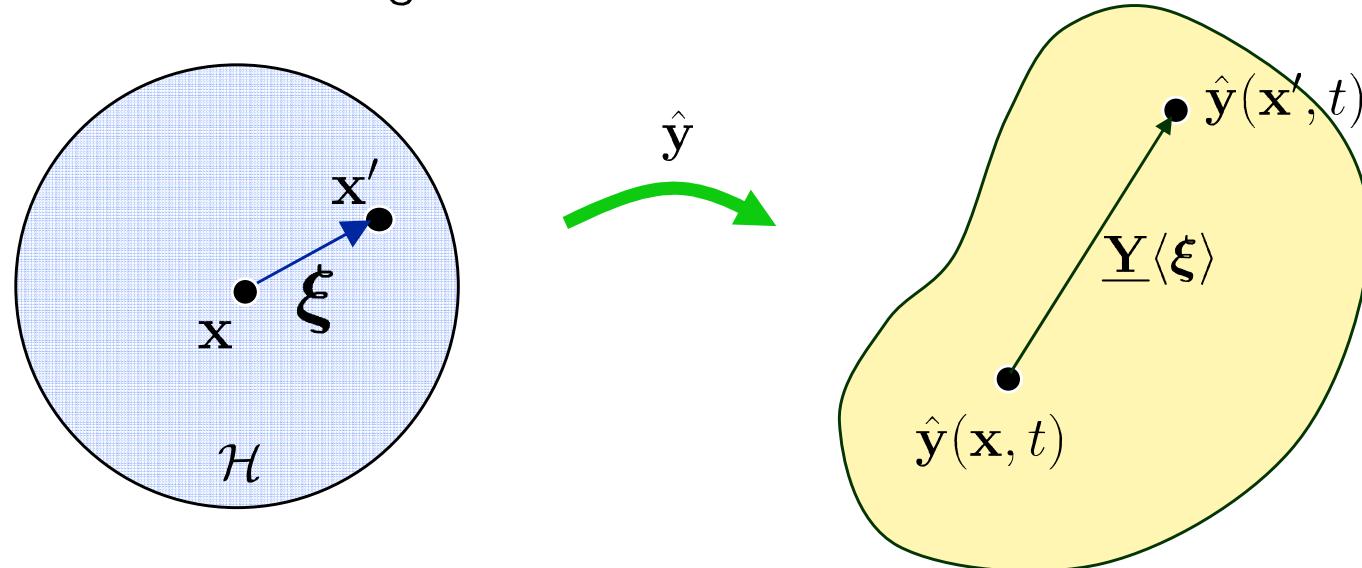
If there is a vector state $\nabla\Psi$ such that

$$d\Psi = \nabla\Psi \bullet d\underline{\mathbf{A}}$$

for any $d\underline{\mathbf{A}}$ then $\nabla\Psi$ is the Frechet derivative of Ψ at $\underline{\mathbf{A}}$.

Deformation states

The deformation state $\underline{\mathbf{Y}}[\mathbf{x}, t]$ maps any bond connected to \mathbf{x} into its deformed image.

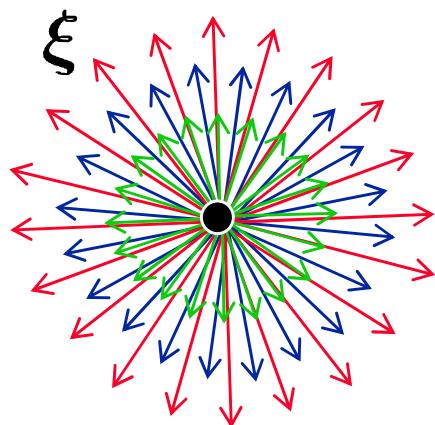


$$\underline{\mathbf{Y}}[\mathbf{x}, t]\langle \mathbf{x}' - \mathbf{x} \rangle = \hat{\mathbf{y}}(\mathbf{x}', t) - \hat{\mathbf{y}}(\mathbf{x}, t)$$

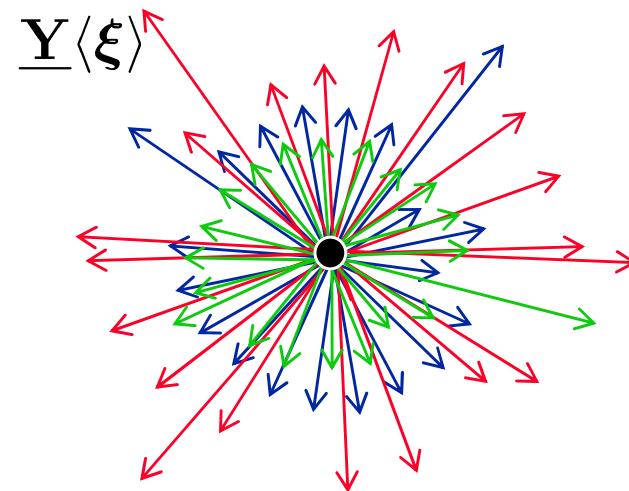


Deformation states contain a lot of kinematical complexity

$\underline{Y}(\xi)$ is the deformed image of any bond ξ .



Undeformed bonds connected to x



Deformed bonds connected to x

Elastic materials

Strain energy density at \mathbf{x} depends only on the deformation state there:

$$W(\mathbf{x}, t) = \hat{W}(\underline{\mathbf{Y}}[\mathbf{x}, t]) \quad (1)$$

Is this really so different from the standard theory?

Standard:

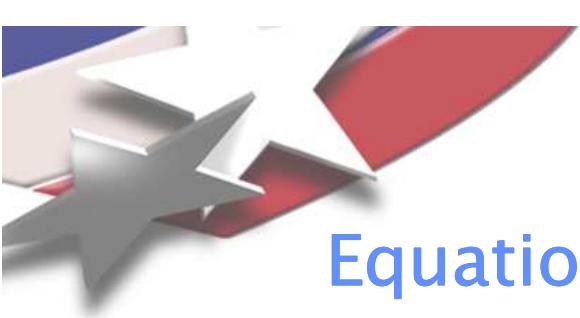
$$\hat{W} \left(\frac{\partial \hat{\mathbf{y}}}{\partial \mathbf{x}} \right)$$

Energy depends on a
linear transformation

Peridynamic:

$$\hat{W}(\underline{\mathbf{Y}})$$

Energy depends on a
nonlinear transformation



Equation of motion from Hamilton's principle

Hamiltonian:

$$H = \int_0^\infty \int_{\mathcal{R}} \left\{ W - \frac{\rho \dot{\mathbf{u}} \cdot \dot{\mathbf{u}}}{2} - \mathbf{b} \right\} dV_{\mathbf{x}} dt$$

Euler-Lagrange equation (equation of motion):

$$\rho(\mathbf{x}) \ddot{\mathbf{u}}(\mathbf{x}, t) = \int_{\mathcal{H}} \left\{ \underline{\mathbf{T}}[\mathbf{x}, t] \langle \mathbf{x}' - \mathbf{x} \rangle - \underline{\mathbf{T}}[\mathbf{x}', t] \langle \mathbf{x} - \mathbf{x}' \rangle \right\} dV_{\mathbf{x}'} + \mathbf{b}(\mathbf{x}, t)$$

where \mathbf{T} is the *force state*:

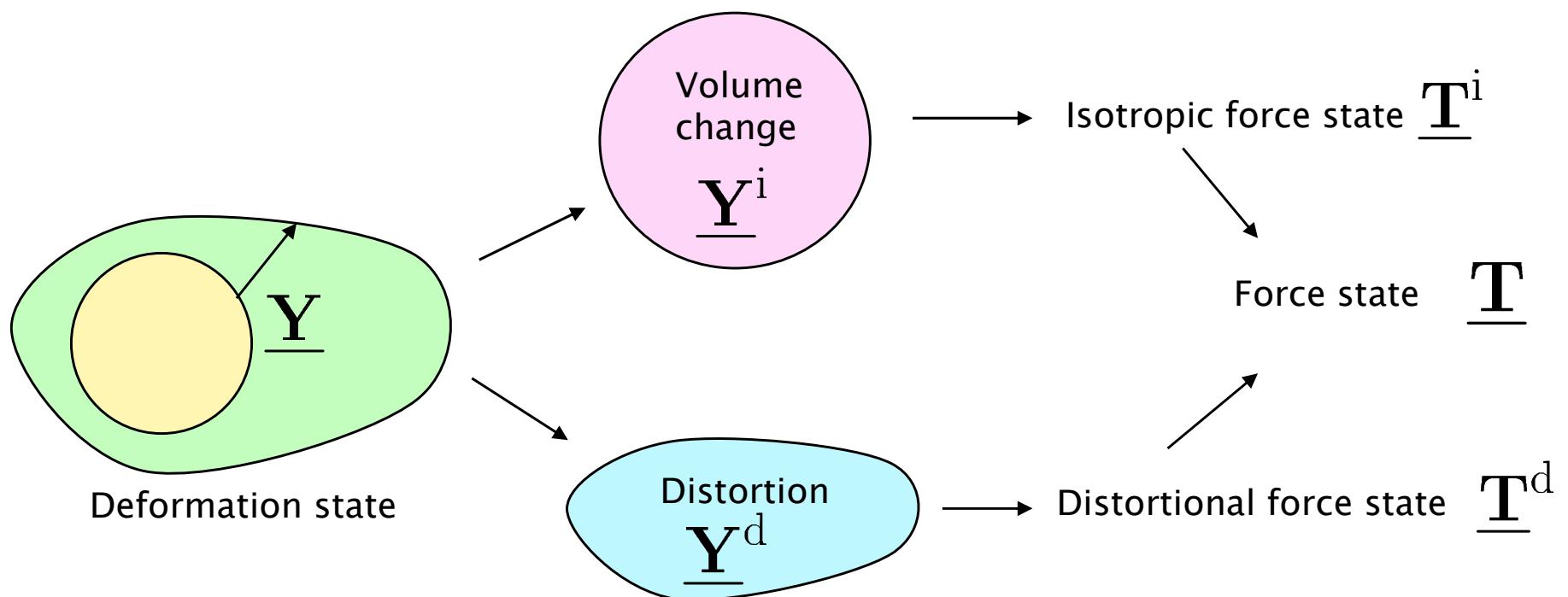
$$\underline{\mathbf{T}}(\underline{\mathbf{Y}}) = \nabla \hat{W}(\underline{\mathbf{Y}})$$

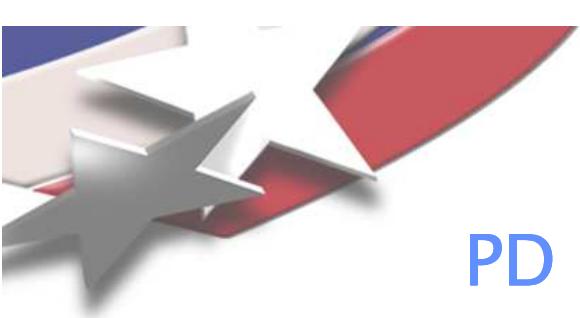
i.e.,

$$dW = \int_{\mathcal{H}} \underline{\mathbf{T}}(\underline{\mathbf{Y}}) \langle \boldsymbol{\xi} \rangle \cdot d\underline{\mathbf{Y}} \langle \boldsymbol{\xi} \rangle dV_{\boldsymbol{\xi}}$$



Can now treat pressure–volume response as decoupled from distortional response





PD material from a conventional model

Define an approximate deformation gradient tensor by

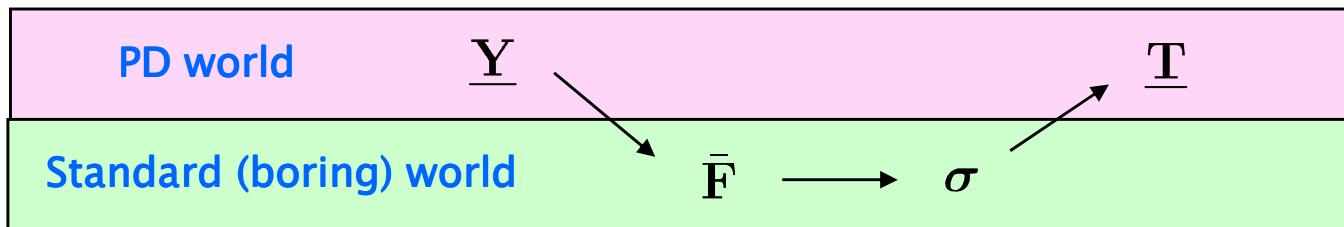
$$\bar{\mathbf{F}}(\underline{\mathbf{Y}}) = m \int_{\mathcal{H}} \underline{\mathbf{Y}} \langle \boldsymbol{\xi} \rangle \otimes \boldsymbol{\xi} \, dV_{\boldsymbol{\xi}} \quad \forall \underline{\mathbf{Y}}$$

Now take any hyperelastic strain energy density function from the standard theory $\Omega(\mathbf{F})$ and set

$$\hat{W}(\underline{\mathbf{Y}}) = \Omega(\bar{\mathbf{F}}(\underline{\mathbf{Y}})) \quad \forall \underline{\mathbf{Y}}$$

Find that the force state is related to the PK stress by

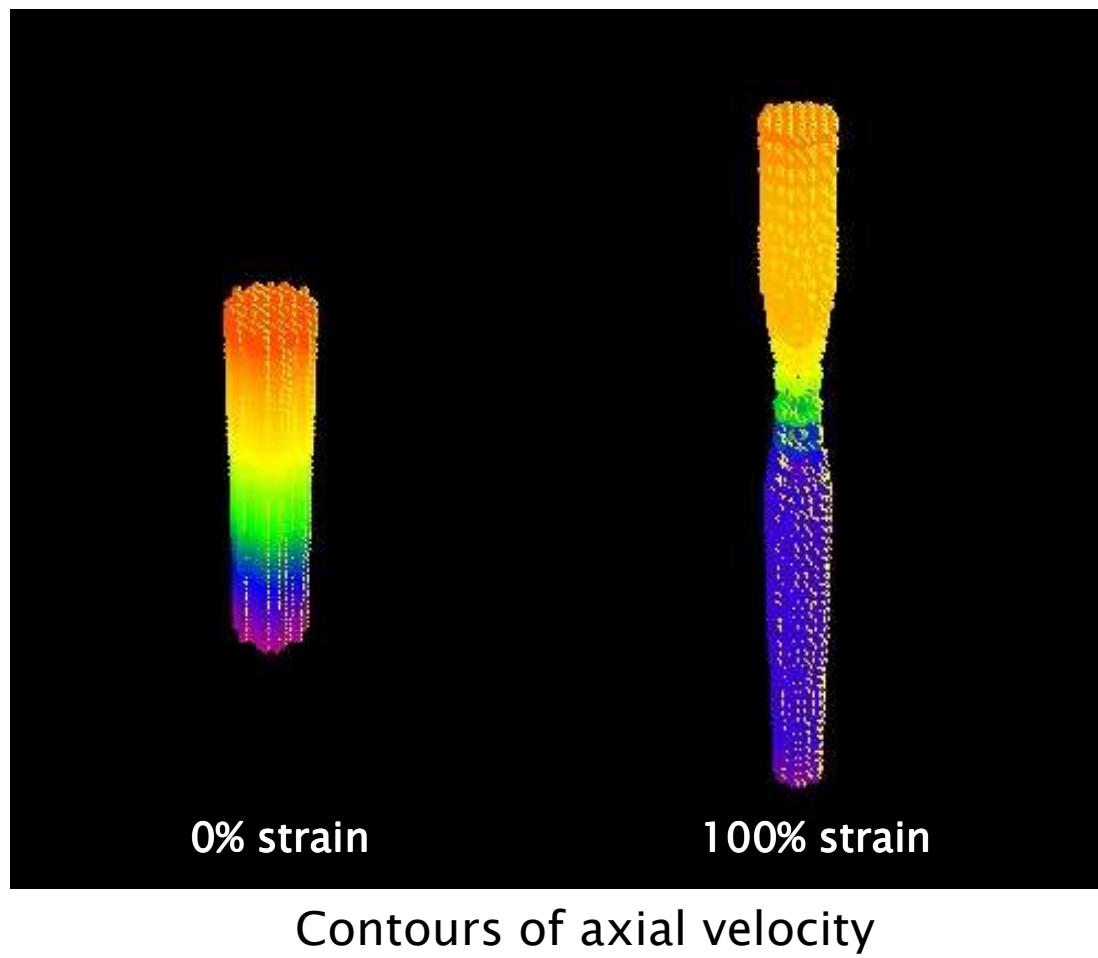
$$\underline{\mathbf{T}} \langle \boldsymbol{\xi} \rangle = \frac{1}{m} \boldsymbol{\sigma} \boldsymbol{\xi}, \quad \boldsymbol{\sigma} = \frac{\partial \Omega}{\partial \mathbf{F}}(\bar{\mathbf{F}})$$





Necking in a 6061-Aluminum Bar

- EMU simulation with state-based peridynamic implementation of large-deformation, strain-hardening, rate-dependent material model.
 - Material model implementation by J. Foster, SNL 5431.



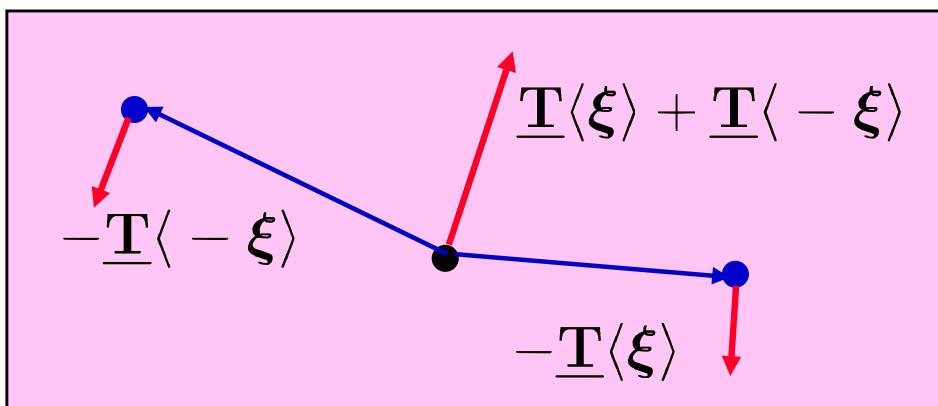
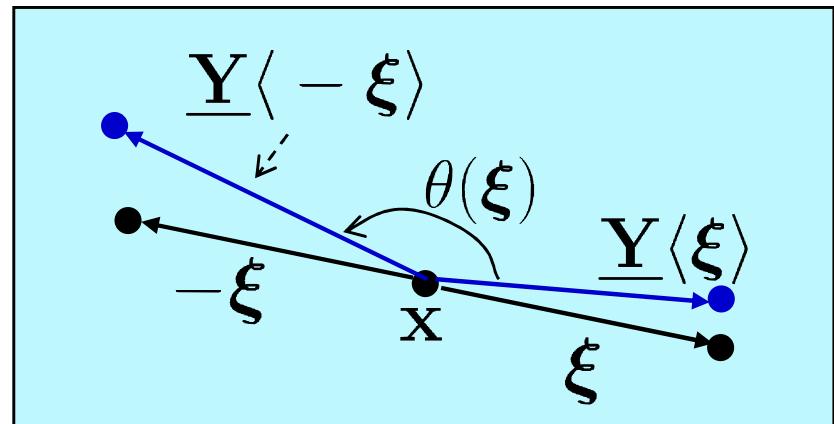


PD states can model interesting collective behavior

Consider a material that responds to angle changes between bonds at $\pm 180^\circ$.

$$\hat{W}(\underline{\mathbf{Y}}) = \int_{\mathcal{H}} (\theta(\xi) - \pi)^2 dV_\xi$$

After evaluating the Frechet derivative, find that the force state resists these angle changes.



Discrete particles and PD states

N -body potential:

$$U(\mathbf{y}_1, \mathbf{y}_2, \dots, \mathbf{y}_N),$$

$\mathbf{y}_1, \dots, \mathbf{y}_N$ = deformed positions, $\mathbf{x}_1, \dots, \mathbf{x}_N$ = reference positions.

Define a PD body by

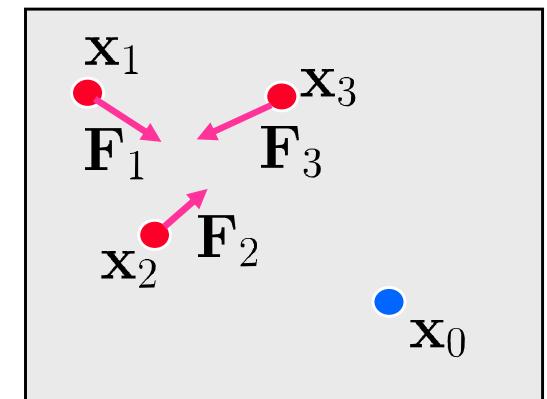
$$\hat{W}(\underline{\mathbf{Y}}, \mathbf{x}) = \delta_d(\mathbf{x} - \mathbf{x}_0) U(\underline{\mathbf{Y}} \langle \mathbf{x}_1 - \mathbf{x}_0 \rangle, \underline{\mathbf{Y}} \langle \mathbf{x}_2 - \mathbf{x}_0 \rangle, \dots, \underline{\mathbf{Y}} \langle \mathbf{x}_N - \mathbf{x}_0 \rangle),$$

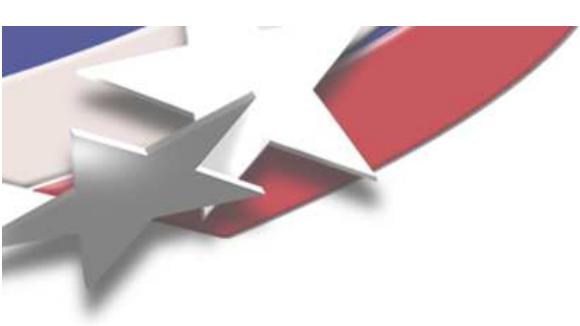
Dirac delta $\rho(\mathbf{x}) = \sum_i \delta_d(\mathbf{x} - \mathbf{x}_i) M_i$ **Particle mass**

where \mathbf{x}_0 is an arbitrary point. Can show the PD equation of motion implies

$$M_i \ddot{\mathbf{u}}(\mathbf{x}_i, t) = -\frac{\partial U}{\partial \mathbf{y}_i}, \quad i = 1, \dots, N$$

Have represented discrete particles as a continuum





Convergence of peridynamics to the standard theory

- Suppose we change the horizon while holding the bulk properties (e.g., bulk modulus) fixed.
- If the deformation is continuous, the kinematical approximation

$$\mathbf{y}(\mathbf{x} + \boldsymbol{\xi}) - \mathbf{y}(\mathbf{x}) = \mathbf{F}(\mathbf{x})\boldsymbol{\xi} + o(|\boldsymbol{\xi}|^2)$$

becomes more and more accurate.

- Therefore

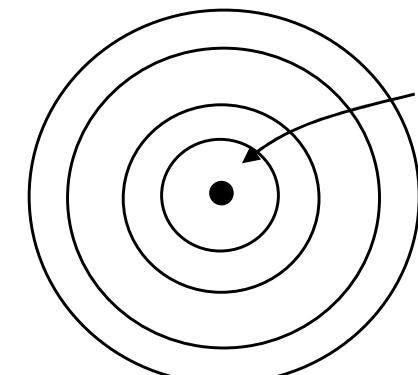
$$W(\underline{\mathbf{Y}}) \approx \hat{W}(\mathbf{F})$$

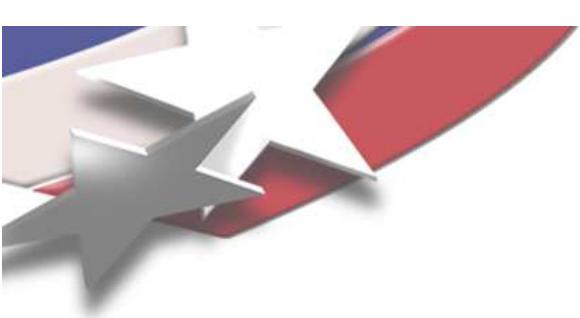
also becomes more accurate.

- The peridynamic force density is closely approximated by

$$\mathbf{L}(\mathbf{x}) \approx \nabla \cdot \boldsymbol{\sigma}, \quad \boldsymbol{\sigma} = \frac{\partial \hat{W}}{\partial \mathbf{F}}$$

Sequence of horizons





Conclusions

- The peridynamic approach offers a consistent mathematical framework for discontinuous problems.
 - Basic equations can be applied directly on cracks.
 - Nonlocality is inherent in the method, length scale is variable.
 - Discrete particles are treated with the same equations as a continuum.