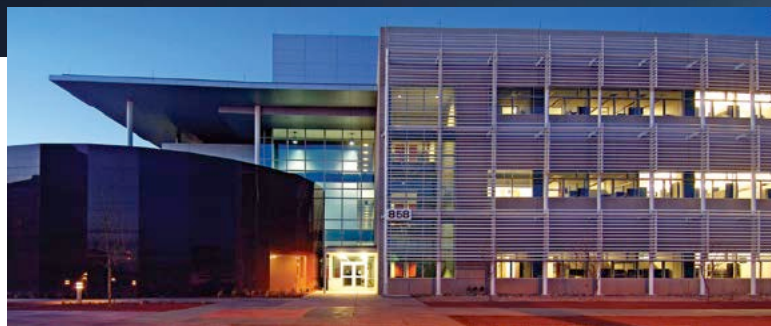


III-Nitride Nanowires: Novel Materials for Lighting and Photovoltaics

Solid-State Lighting Science Energy Frontier Research Center
Sandia National Laboratories, Albuquerque, NM

George T. Wang*, Qiming Li, Jonathan J. Wierer, Daniel D. Koleske, Jeffrey J. Figiel,
Jeremy B. Wright, Ting S. Luk, Igal Brener



Sandia MESA Facility



e-mail: gtwang@sandia.gov

Sandia Albuquerque



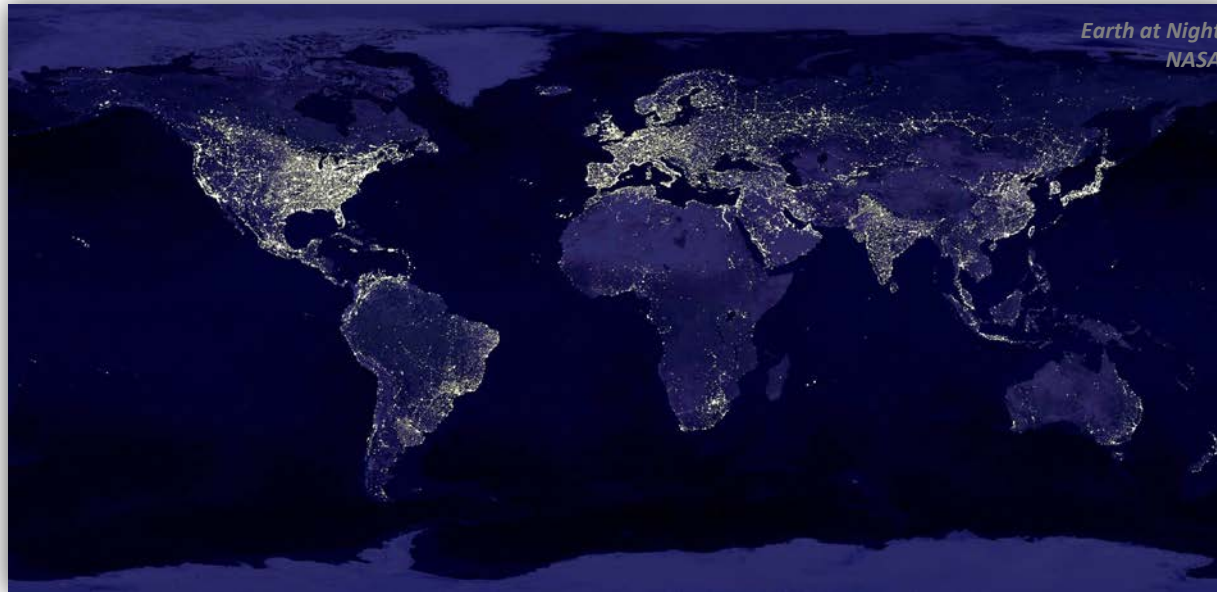
Outline

- **Solid-state lighting (SSL) overview**
- **Why nanowires for SSL?**
- **Bottom-up nanowires**
 - **Growth, characterization, nanowire-templated film growth**
- **Top-down nanowires**
 - **Fabrication, axial and radial LEDs, solar cell, lasers**
- **Summary**





Lighting is a large fraction of energy consumption and is low efficiency



Efficiencies of energy technologies in buildings:

Heating:	70 - 80%
Elect. motors:	85 - 95%
Fluorescent:	~17%
Incandescent:	~4%

Lighting is one of the most inefficient building energy technologies → opportunity!

~22% of US electricity use is for general illumination (~1/15 world's energy, \$330B in 2005)

Current SSL efficiency: ~20%
US DOE target: 50%
"Ultra-efficient" SSL*: $\geq 70\%$



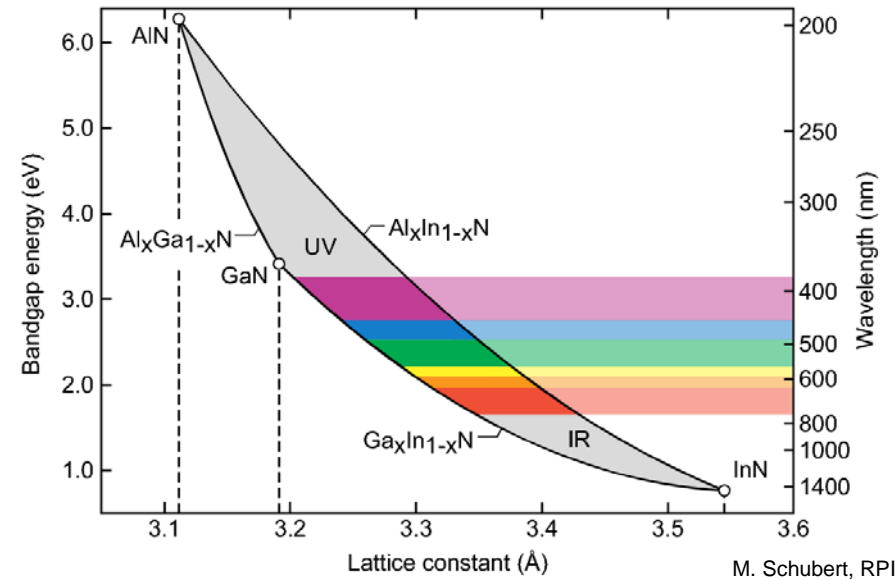
2011 Philips L-prize 60W equivalent (940 lumens) 10W LED light bulb replacement
<http://www.lightingprize.org/philips-winner.stm> \$49 (as of May 2012)

Achieving 50% or greater efficient lighting would have tremendous global impact!



Foundation of SSL: III-Nitride (AlGaInN) Semiconductors

- **Direct RT bandgaps: ~0.7-6.2 eV**
- Solid alloy system (tuneable bandgaps)
- High breakdown field, mobility, thermal conductivity, melting temperature
- Radiation resistant and chemically inert
- InGaN covers entire visible & bulk of solar spectrum (PV material?)



- *Used in LEDs, blue laser diodes, high power transistors, HEMTs*

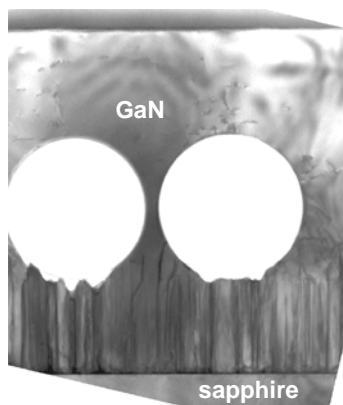
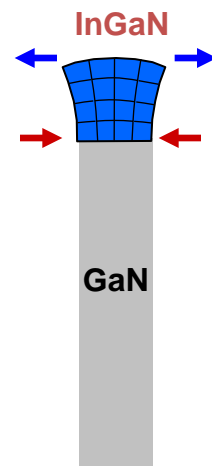


Nitronex GaN power transistor

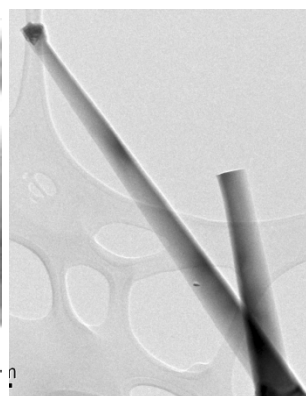
Why III-nitride (AlGaInN) nanowires for SSL, PV, etc.?

Strain accommodation

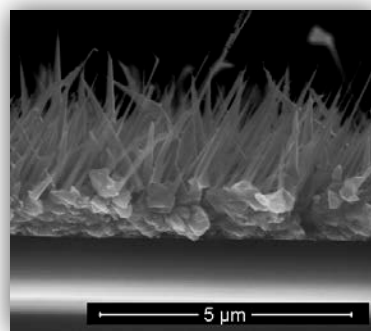
Nanowire attribute	SSL Benefit
Few or no threading dislocations	IQE, device lifetimes
Growth on variety of substrates	Cost, flexibility
Reduced piezoelectric strain	IQE (less QCSE)
Wider range of alloy compositions possible	Wider color space (green-yellow-red)



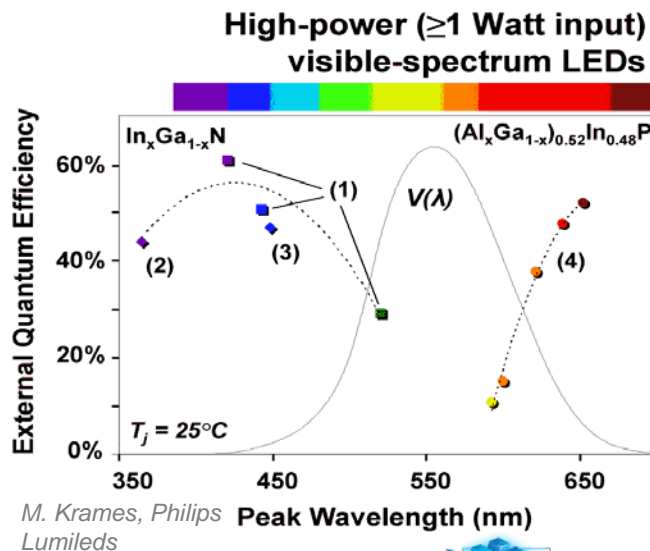
GaN on sapphire
(SiO₂ spheres for
dislocation blocking)



Dislocation-free GaN NWs



GaN NWs on tungsten foil

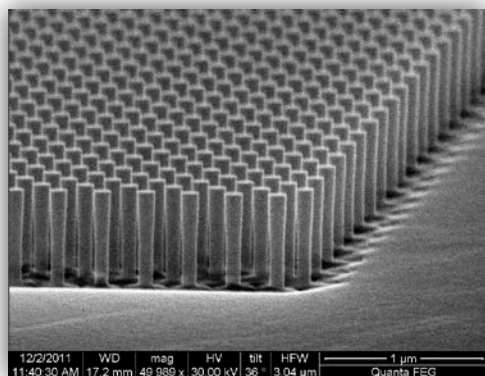




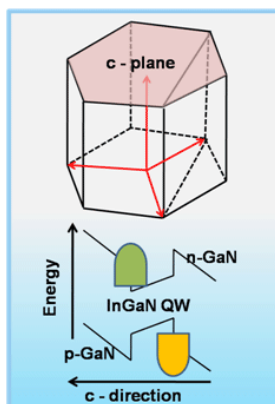
Why III-nitride (AlGaInN) nanowires for SSL, PV, etc.?

Size, geometry

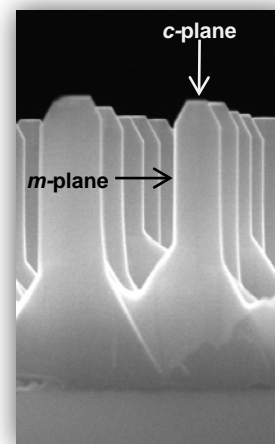
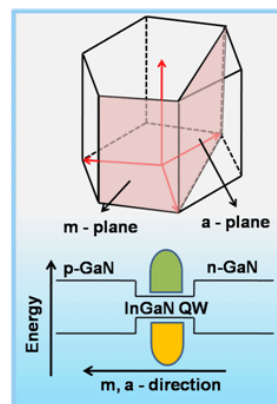
Nanowire attribute	SSL Benefit
2D arrangements, e.g. photonic crystals	Enhanced/controlled light extraction, enhanced radiative recombination (IQE)
Vertical 3D device integration	High surface/device area = lower cost;
Non-polar, semi-polar facets	Reduced polarization effects (IQE)
Lasers	Lower threshold lasers; reduced eff. droop
Discrete; all of the above	Scientific test system



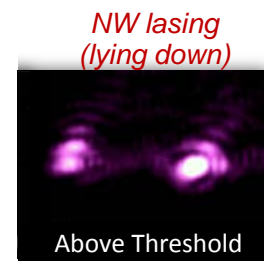
GaN NW photonic crystal



Non - polar direction GaN thin film growth → higher quantum efficiency




GaN/InGaIn MQW NW

<http://cse1.snu.ac.kr/research/LED.php>

Bottom-up III-Nitride Nanowires

Research Highlights

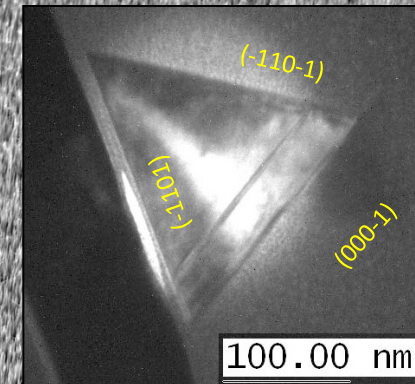
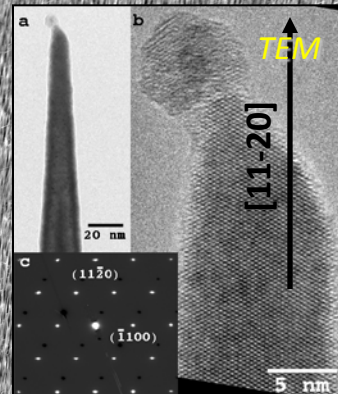
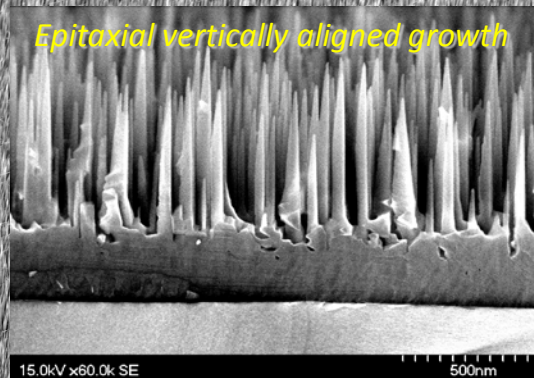


A scanning electron micrograph (SEM) showing a dense array of vertical nanowires. The nanowires are uniform in height and spacing, growing from a substrate. A scale bar at the bottom right indicates a length of 100 micrometers.

100 μm



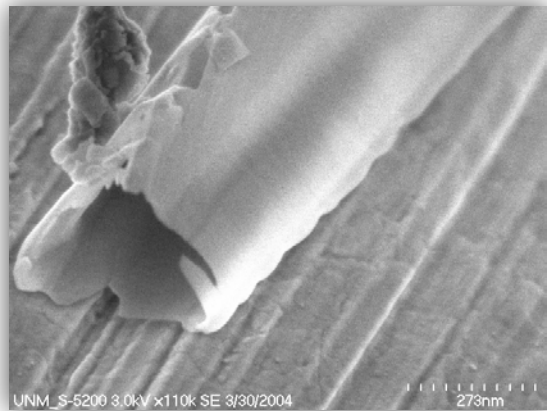
“Bottom Up” aligned GaN nanowire growth



- Nanowires grown by Ni-catalyzed MOVPE/MOCVD (VLS)
- Highly-aligned vertical growth over large areas (2" r-sapphire wafer)
- Controllable densities as high as ~ 150 nanowires μm^{-2}
 - Q. Li, G. T. Wang, *Appl. Phys. Lett.* 93, 043119 (2008)
 - Q. Li, J. R. Creighton, G.T. Wang *J. Cryst. Growth* 310 3706-3709 (2008)
- Primary [11-20] growth orientation (\perp to (11-20) α -plane)
- Triangular faceted -- (000-1) and equiv. (-1101) and (-110-1)
- TEM: Single crystal, dislocation free; c-plane stacking faults
 - G. T. Wang et al., *Nanotechnology* 17 5773-5780 (2006)

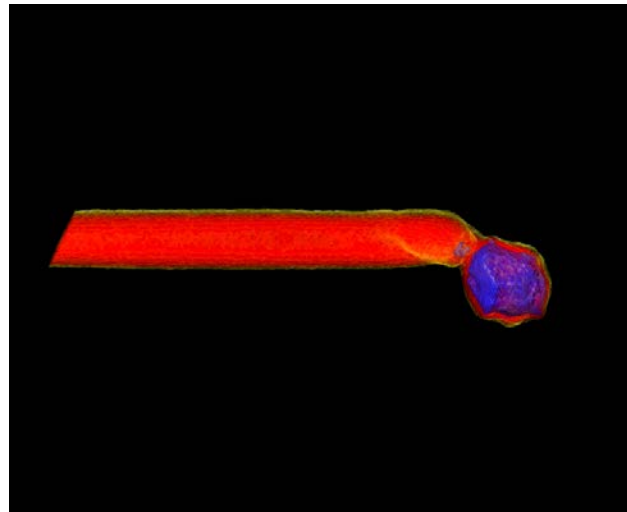


Radial heterostructure nanowire growth

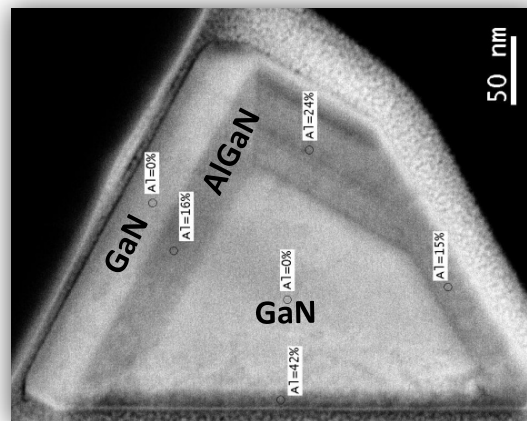
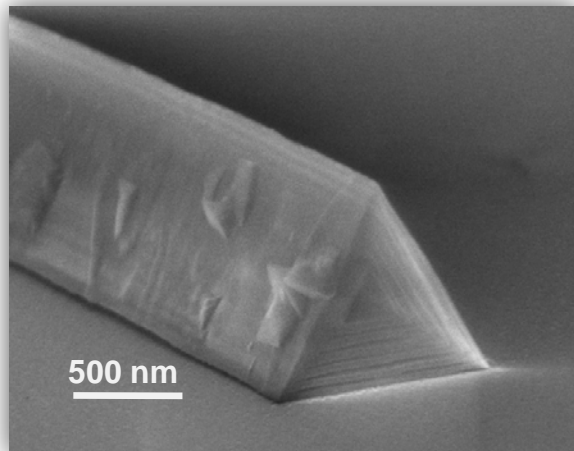
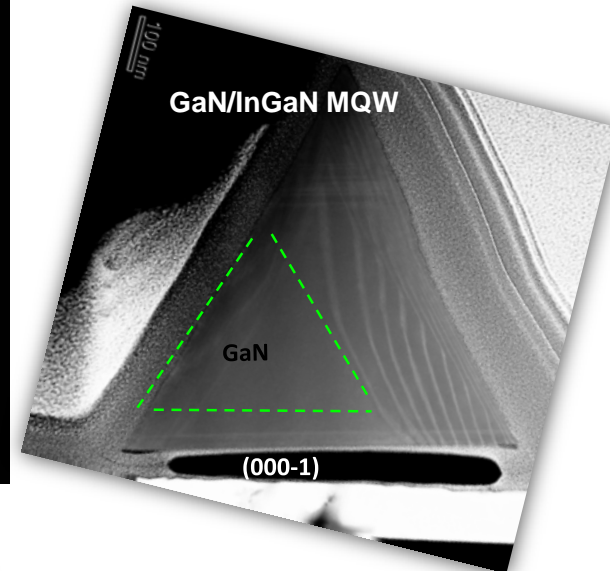


G. Stan et al., *Nanotechnology*, **20**, 2009

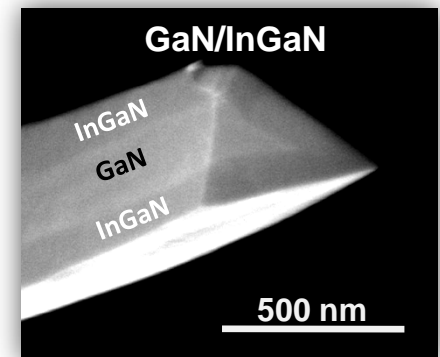
AlN Nanotubes



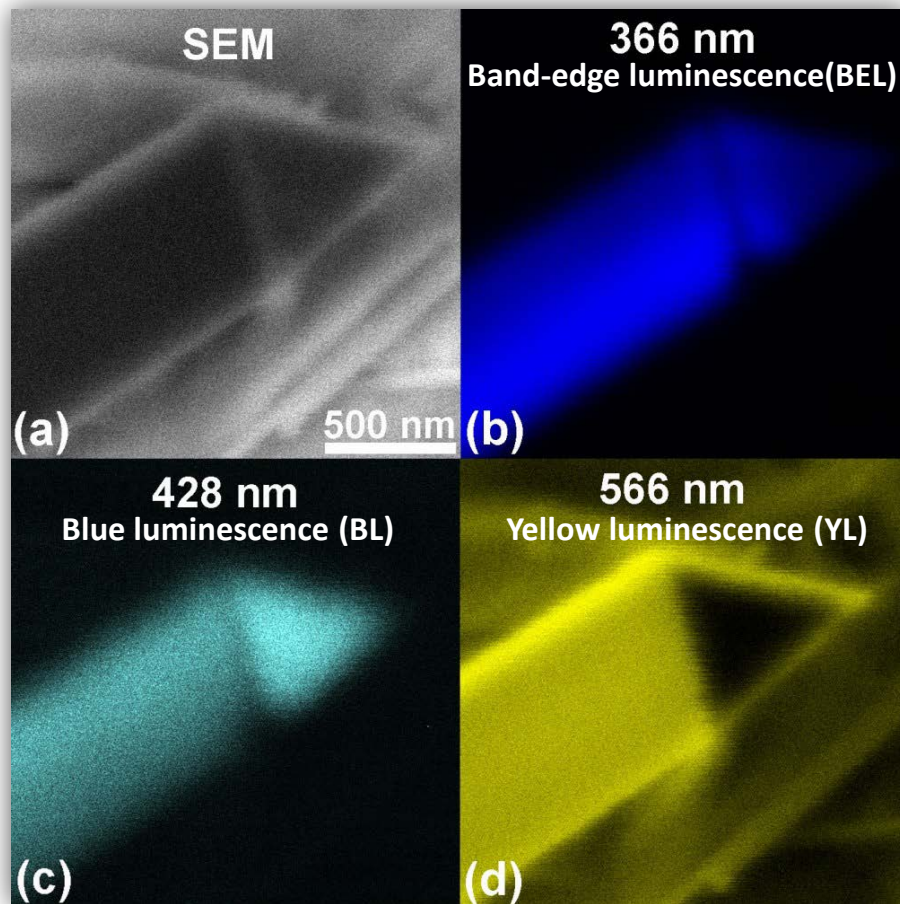
GaN/AlN NW



GaN/AlGaIn/GaN
Double heterostructure



Spatial distribution of luminescence in GaN NWs



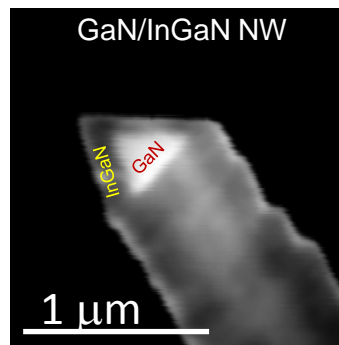
- Band-edge luminescence (BEL) at ~ 366 nm and defect-related blue luminescence (BL) at ~ 428 nm observed in NW core/bulk
- **Defect-related yellow luminescence (YL) exhibits strong surface component** -- associated with surface states or concentrated near surface region
- YL in GaN attributed to many possible sources (C, O impurities, Ga vacancies, etc.)
- Isolated Ga vacancies have low diffusion barrier (~ 1.5 eV) & may migrate toward surface during growth
- BL linked to $V_{\text{Ga}}\text{-O}_{\text{N}}$ ($D \sim 2.2$ eV), less mobile

Nanoscale Cathodoluminescence (CL) imaging: Cross-section GaN NW

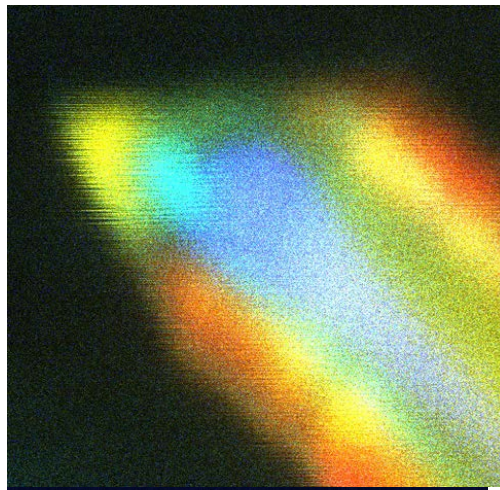
Q. Li, G. T. Wang, *Nano Lett.*, 2010, 10 (5), 1554

Highlight: Core-shell NWs – A good platform for high In InGaN?

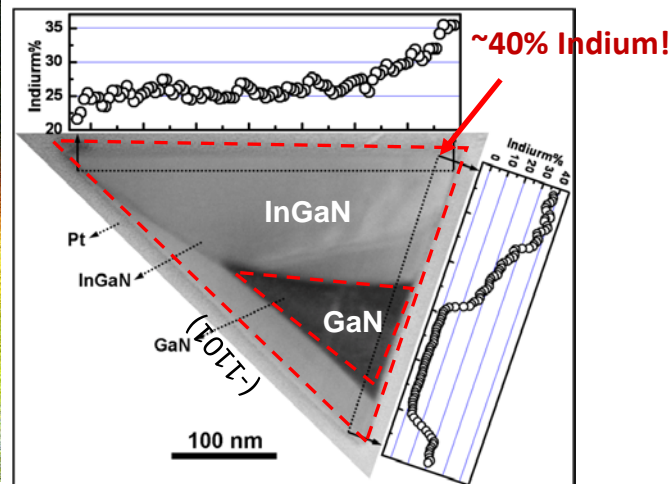
- **Issue: Strain limits In incorporation in InGaN thin films (green-yellow-red gap)**
- Radial core-shell NWs: much higher active region area than axial NW or planar heterostructures



GaN core – 900 °C, 10 min.
InGaN shell – 760 °C, 60 min.

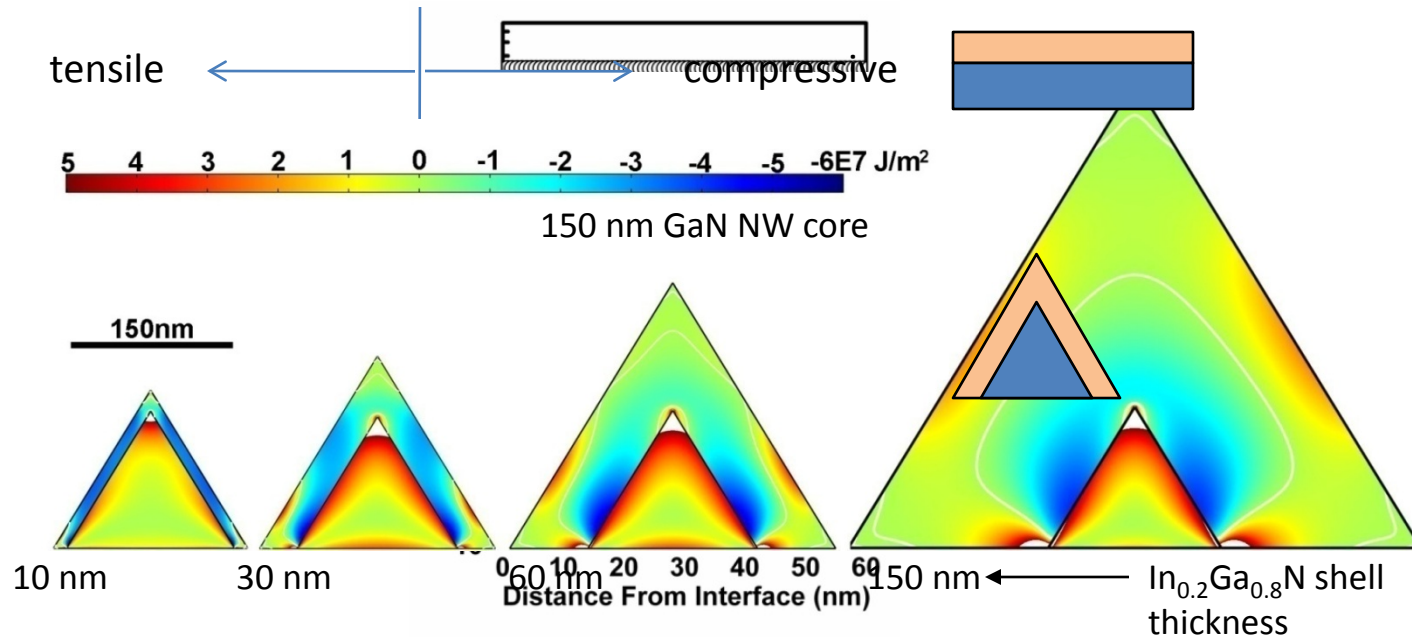


- CL shows strong multicolor emission up into the IR!
- **high In incorporation with high material quality**
 - nonuniform In composition?



STEM/EDS shows In distribution, **highest at surface/corners**

Reduced strain in nanowires allows higher In incorporation

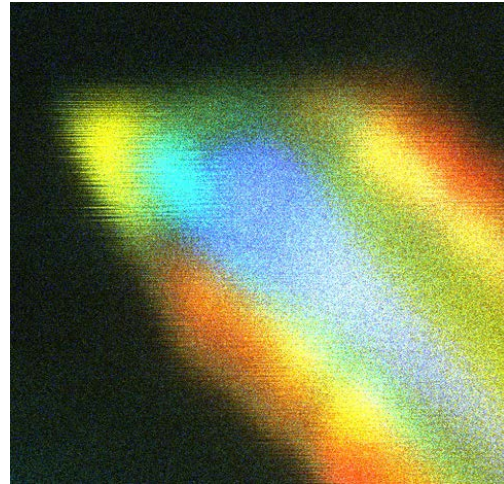
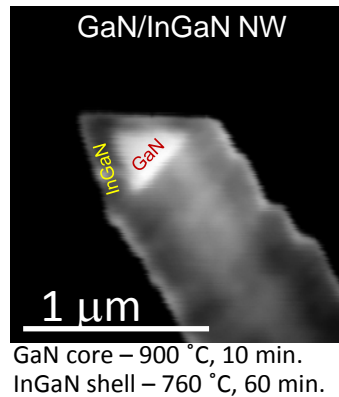


- Strain lower for GaN/InGaN nanowire than for GaN/InGaN thin film
- Compressive strain decreases away from GaN/InGaN interface, lowest in corners

Q. M. Li, G. T. Wang, "Appl. Phys. Lett.", **97**, 181107 **2010**.

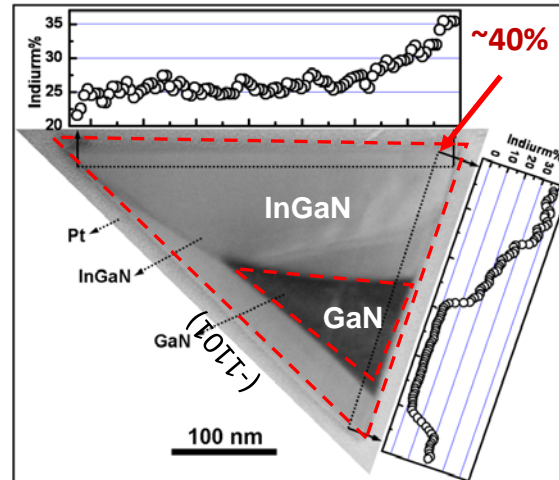
Highlight: High In incorporation in GaN/InGaN core-shell NWs

- **Issue: Strain limits In incorporation in InGaN thin films (green-yellow-red gap)**
- Radial core-shell NWs: much higher active region area than axial NW or planar heterostructures

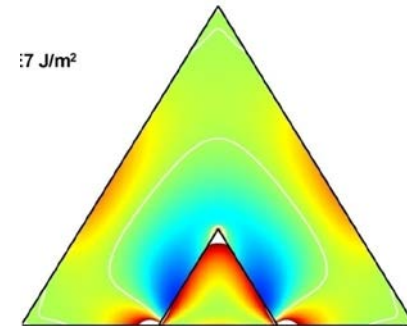


CL shows strong multicolor emission up into the IR!

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STEM/EDS shows In distribution, **highest at surface/corners**



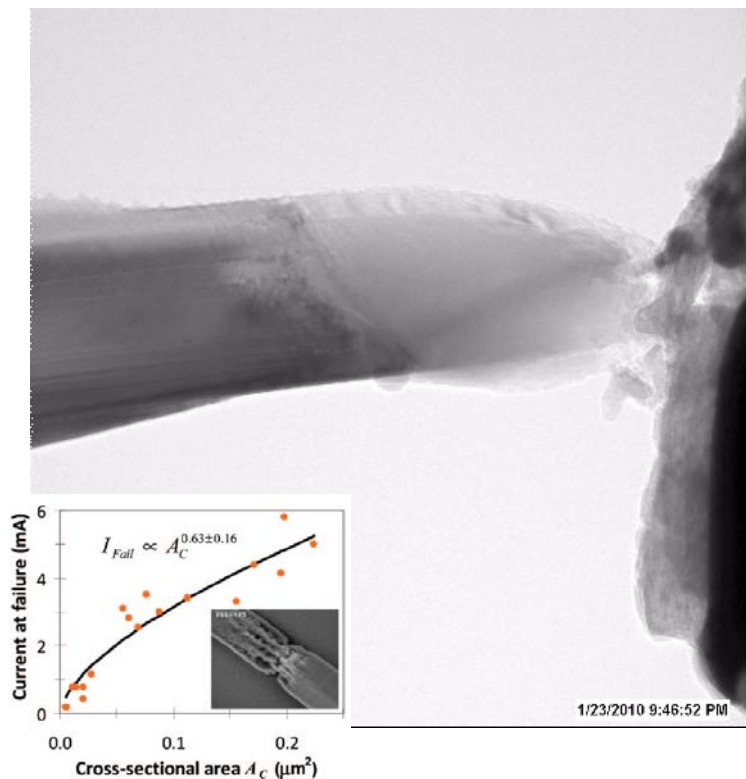
2D FEA: highest In regions correlated w/lowest compressive strain regions

Q. M. Li, G. T. Wang,
“Appl. Phys. Lett.”, **97**,
181107 **2010**.

Radial InGaN/GaN nanowires promising for addressing green-yellow-red gap

In-situ TEM studies

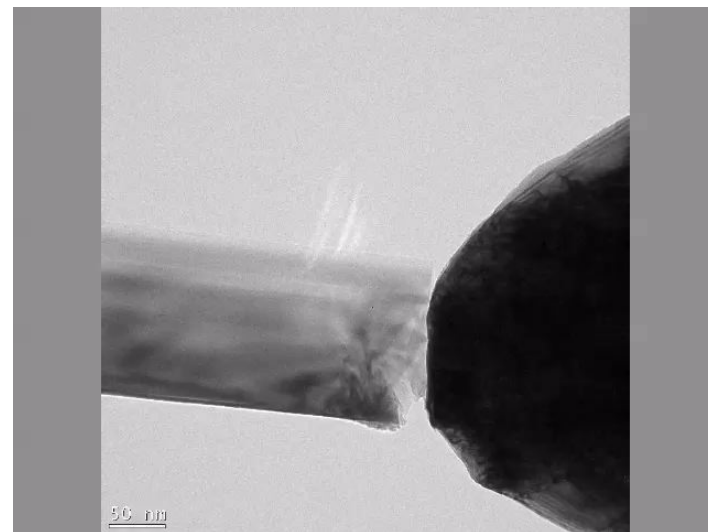
GaN NW decomposition via Joule heating
(relevant for NW devices)



NW breakdown at 60V, 20 μA
(avg. breakdown $I \sim 3000 \text{ kW/cm}^2$)

T. Westover, R. Jones, J. Y. Huang, G. Wang, E. Lai, A. A. Talin, *Nano Lett.*, **9**, 257 (2009).

Local plastic deformation of GaN NW
(relevant for NW devices)



- Dislocation-free NW shows significant surface plastic deformation
- Mediated by dislocation nucleation & pile up, grain boundary sliding

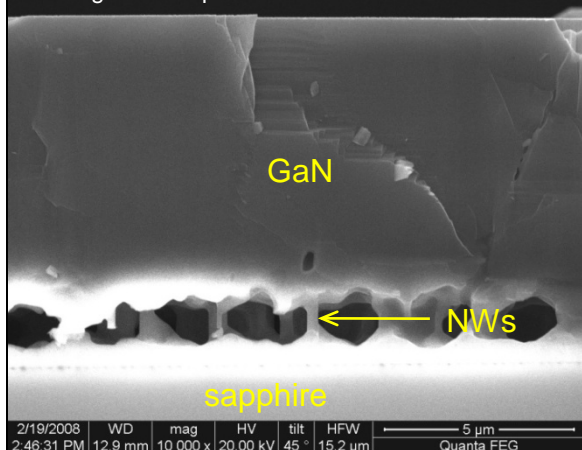
J. Y. Huang, H. Zheng, S. X. Mao, Q. Li, and G. T. Wang, *Nano Lett.*, **11** (4), 1618 (2011).



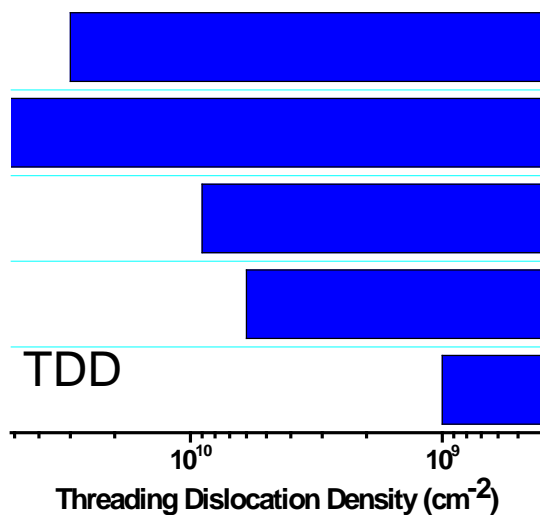
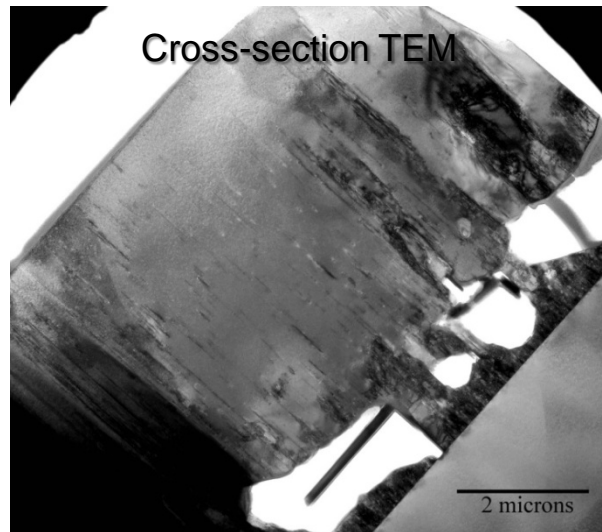
Nanowire-templated lateral epitaxial growth (NTLEG) of GaN

Inexpensive method to reduce dislocation density in GaN films growth on lattice mismatched substrates

Nanowire array growth: 780 °C 3 min.
Lateral growth step 1: 900 °C 15 min.
Lateral growth step 2: 1075 °C for 15 min.



Cross-section SEM



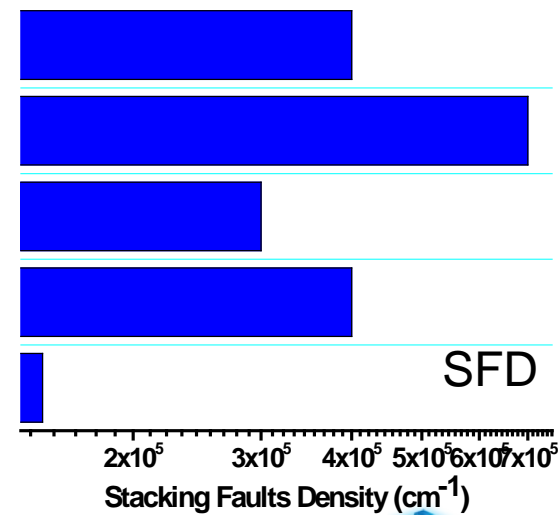
Craven et al., *APL* 81(2002), 469
(LT nucl. Layer)

Chakraborty et al. *APL* 89(2006), 041903
(LT nucl. Layer)

Chakraborty et al. *APL* 89(2006), 041903
(SiNx Nanomask)

Qian et al, *JAP* 106(2009), 123519
(HT AlN nucl. Layer and 3-step growth)

Li et al. *Adv. Mat.*, 21(2009), 2416
(NTLEG) THIS WORK

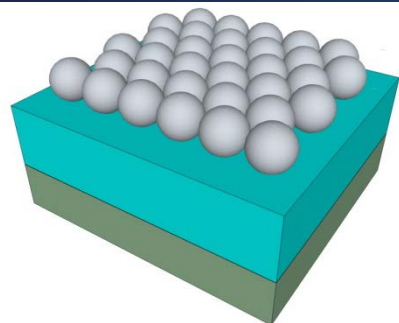


The background of the slide is a scanning electron micrograph (SEM) showing a dense array of vertical nanowires. The nanowires are uniform in height and width, standing on a textured substrate. The image is in grayscale, with the nanowires appearing as bright, vertical lines against a darker, more detailed background.

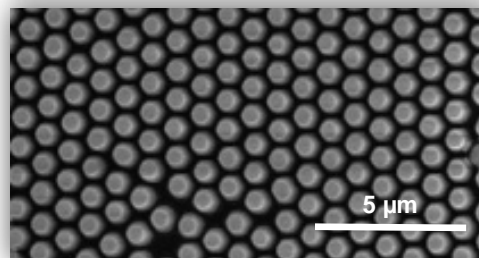
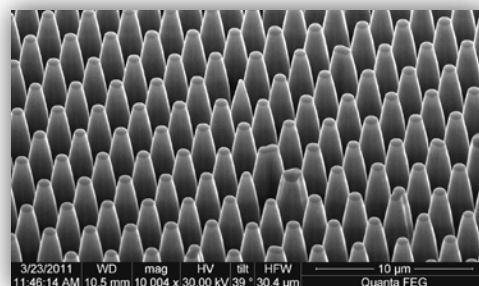
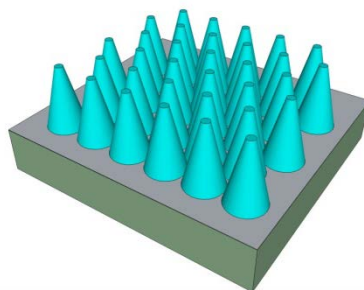
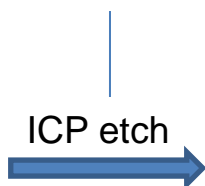
Top-down III-Nitride Nanowires

Research Highlights

New dry + wet top-down ordered nanowire fabrication process

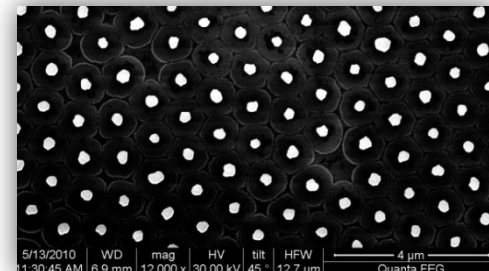
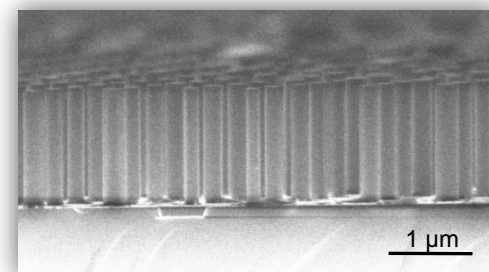
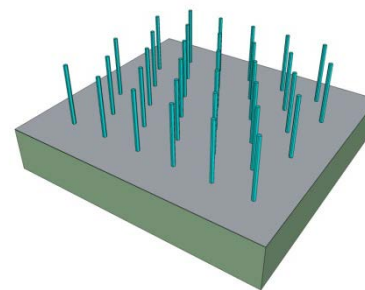


(0001) GaN on sapphire

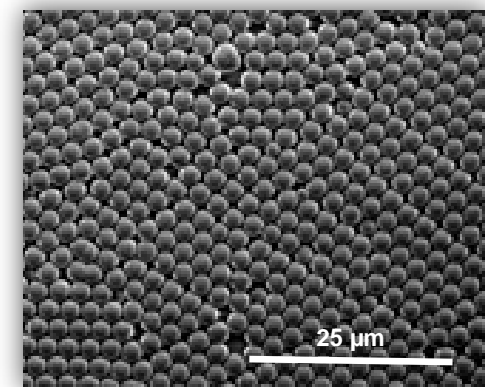


Plasma etch causes sidewall damage
C. Y. Wang et al., Opt. Expr. 16, 10549–10556, 2008.
Tapered; no well-defined facets

Selective wet etch
(AZ-400K developer)



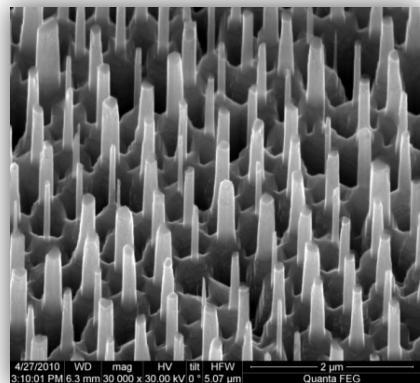
Wet etch: straight sidewalls,
removes sidewall damage



Q. Li, J. J. Figiel, G. T. Wang, Appl. Phys. Lett., **94**, 231105 (2009).

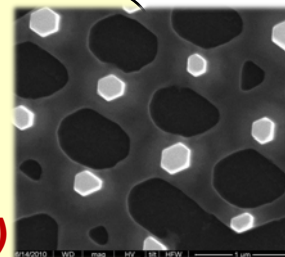
Straight GaN nanowires with controllable geometries

0.5 μm sphere size

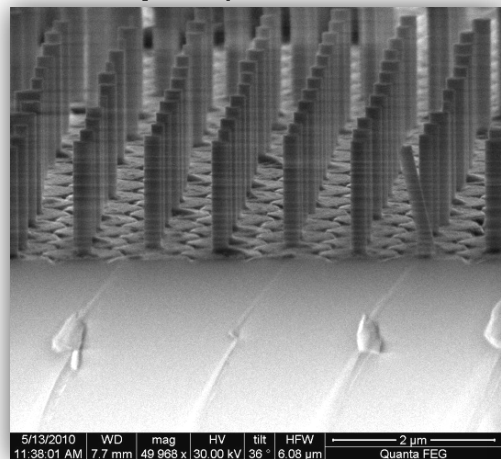


Wet etch rate negligible for top c-face & fast for $\{10\bar{1}0\}$ *m*-facets, leads to hexagonal NWs with **straight & smooth** *m*-facets

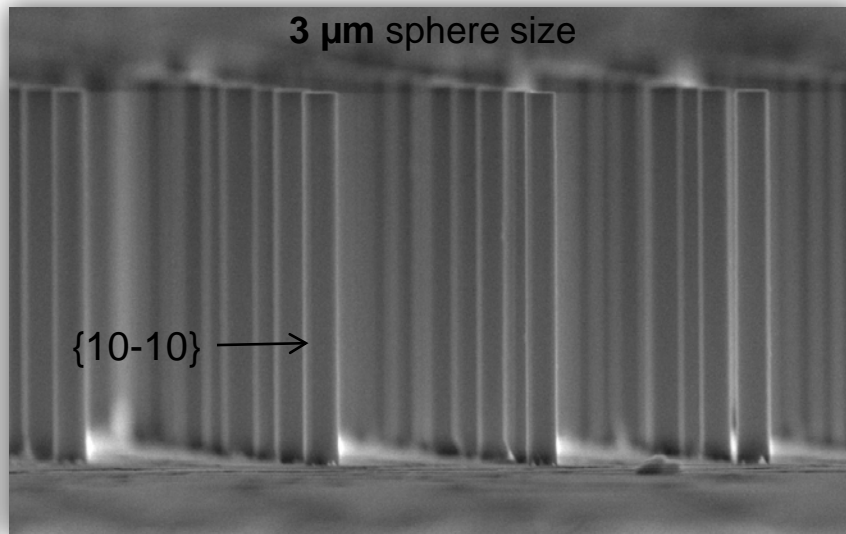
New fabrication method allows for superior control of:
Height (dry etch depth)
Diameter (wet etch time)
Pitch/arrangement (defined by masking template)



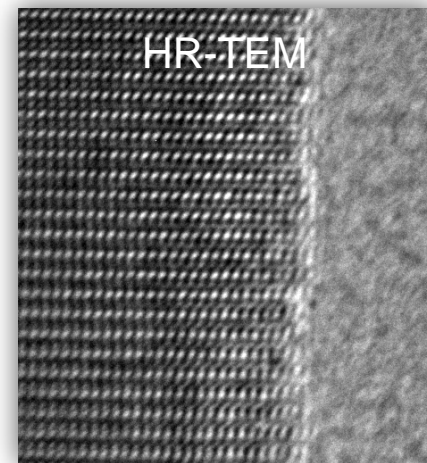
1 μm sphere size



3 μm sphere size



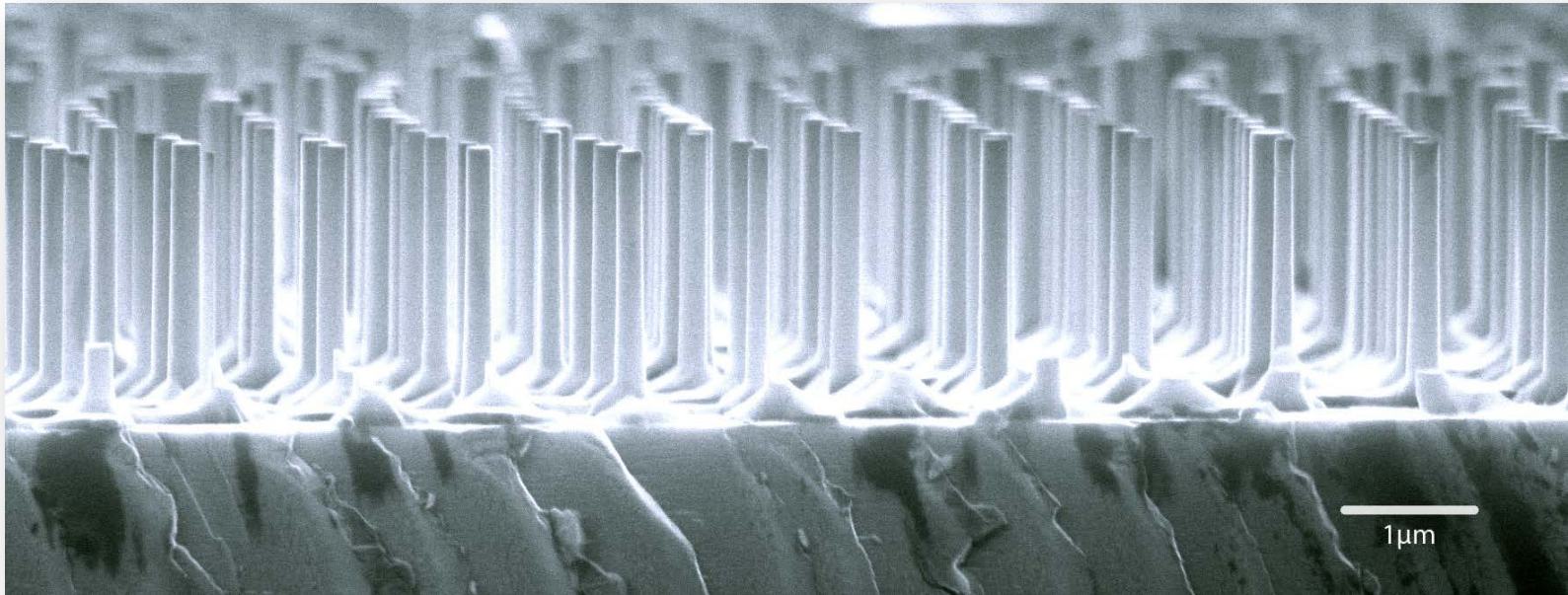
HR-TEM



Smooth sidewall created by wet etch

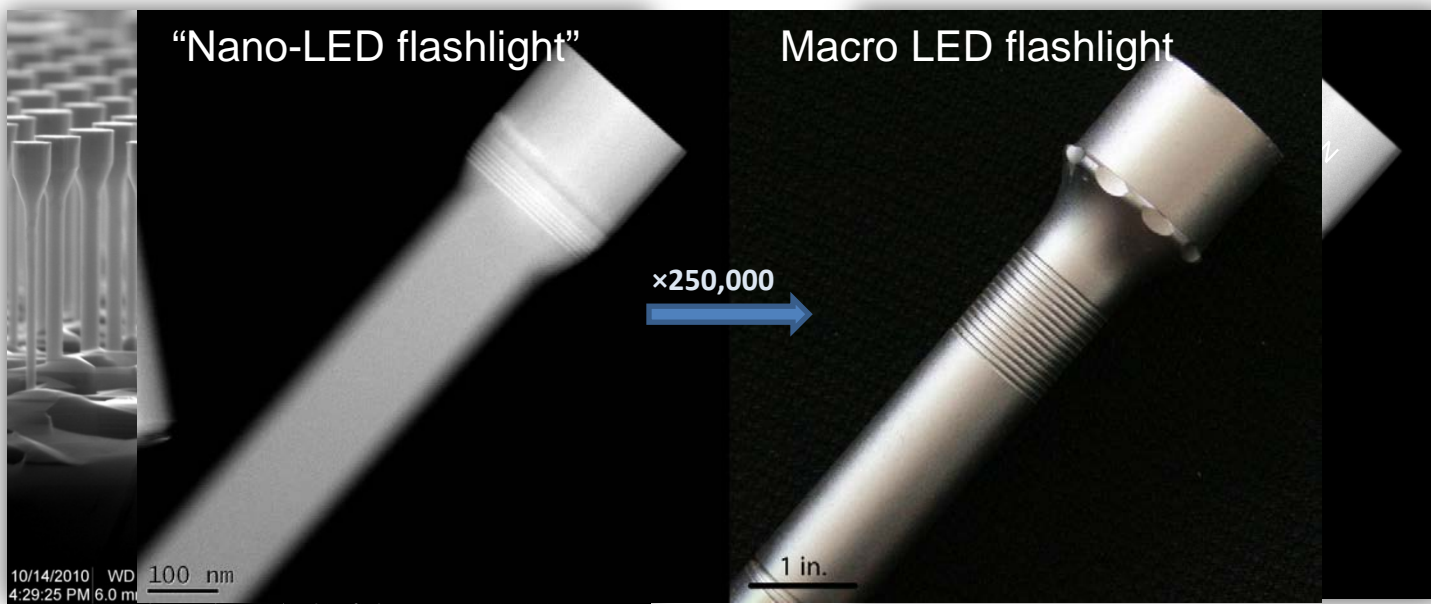
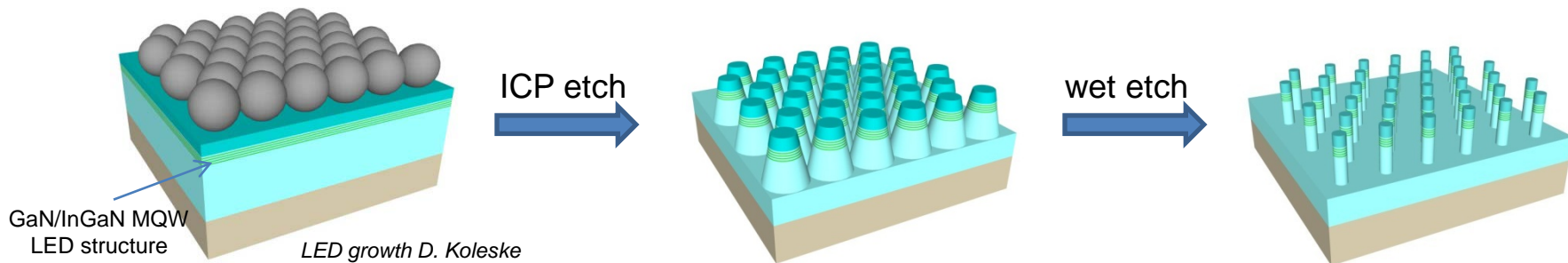


Advantages of new top-down nanowire fabrication method



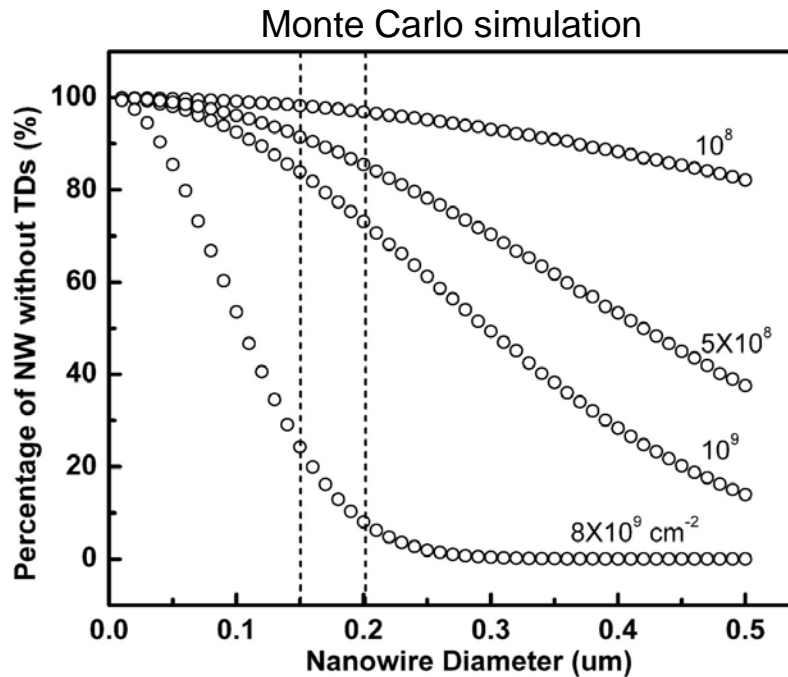
- Wider range of growth conditions, material tunability (1D growth cndns not needed)
- Lower point defect densities (higher growth temperature)
- Ordered/periodic arrays (difficult with catalyst/VLS-based methods)
- Axial III-nitride nanowire heterostructures possible by MOCVD
- Better control of geometry (independent control over height, width, & pitch)
- Improved uniformity
- Easier vertical device integration (height uniformity, base GaN-layer)

Highlight: Axial GaN/InGaN nanowire LEDs

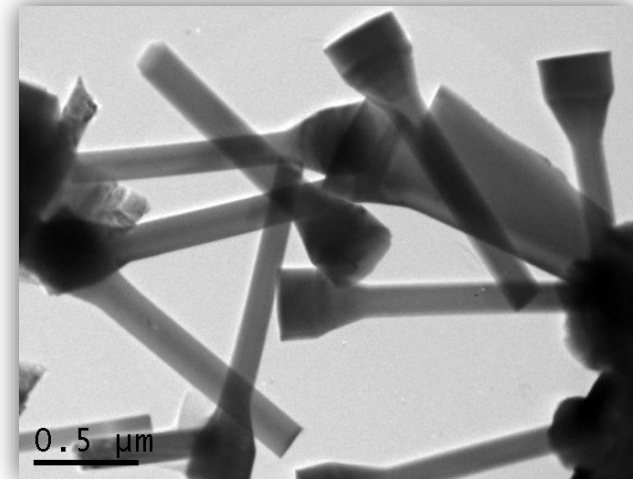


Q. Li et al.,
Optics Express **19**,
25528 (2011)

Nanorod threading dislocations as function of diameter



Bright-field TEM



Nanorods etched from $\sim 5 \times 10^8 \text{ cm}^{-2}$ planar LED

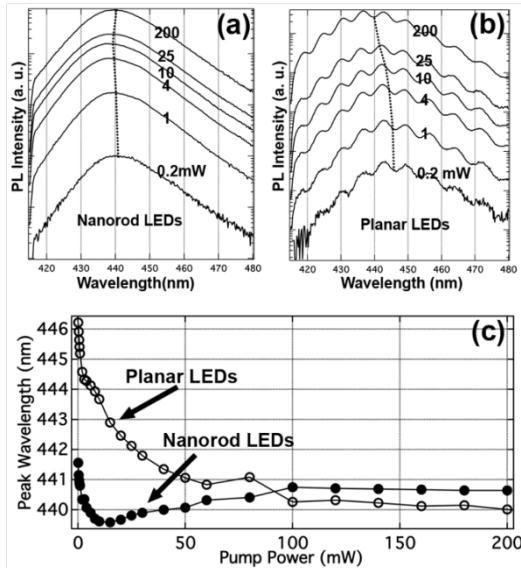
- Etched nanorods inherit the dislocation density of the parent film
- However, as the diameter approaches zero, the *fraction* of nanorods with one or more dislocations also approaches zero! [$\# \text{ TDs per rod} \sim (\text{TDD}) \times (A_{\text{cross-section}})$]
- *~94% of nanorods ~150 nm in diameter from TDD ~ $5 \times 10^8 \text{ cm}^{-2}$ film dislocation free!*

Optical performance – axial nanowire LEDs vs. planar LEDs

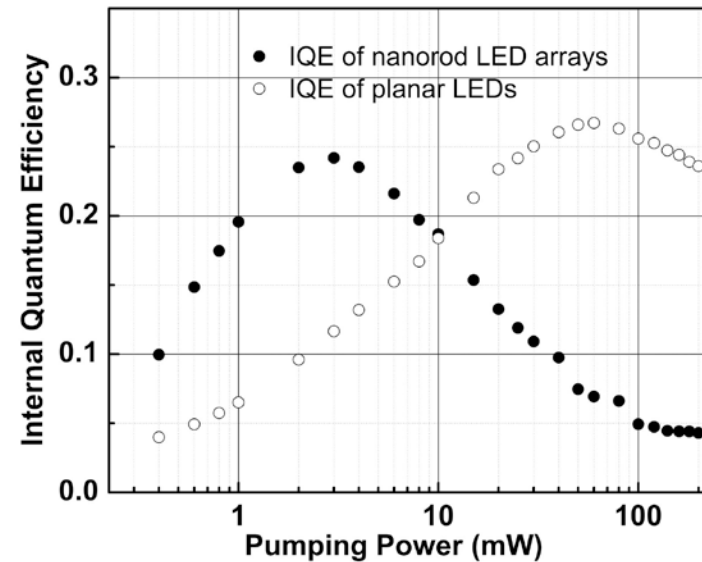
413 nm pump (InGaN selective)

PL, IQE measurement K. Westlake, M. Crawford

InGaN peak position vs pump power



IQE – nanowire vs Film



- XRD shows $\sim 16 \pm 4\%$ strain reduction in InGaN QWs in nanowire LEDs
XRD measurement courtesy Steve Lee
- Little wavelength shift at higher pump powers for nanowire LEDs (no/reduced QCSE)

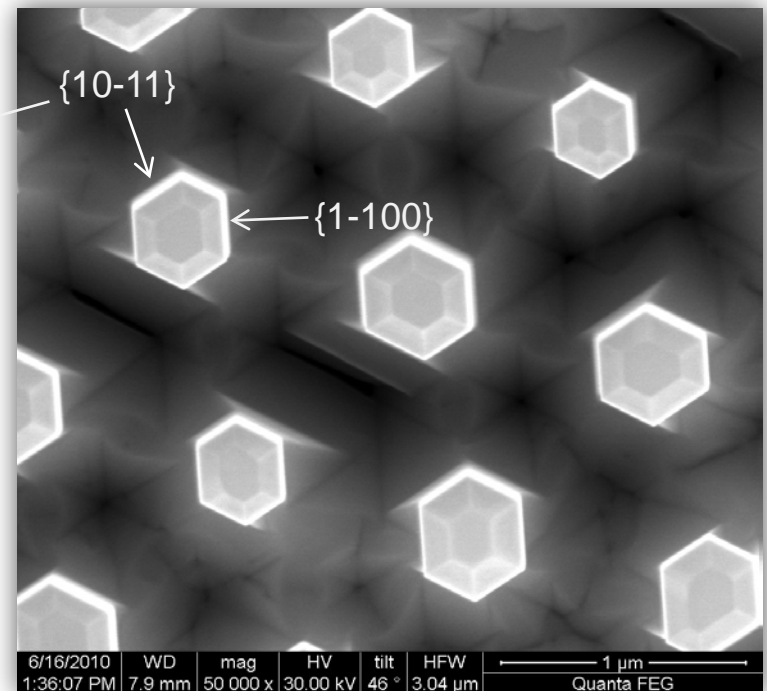
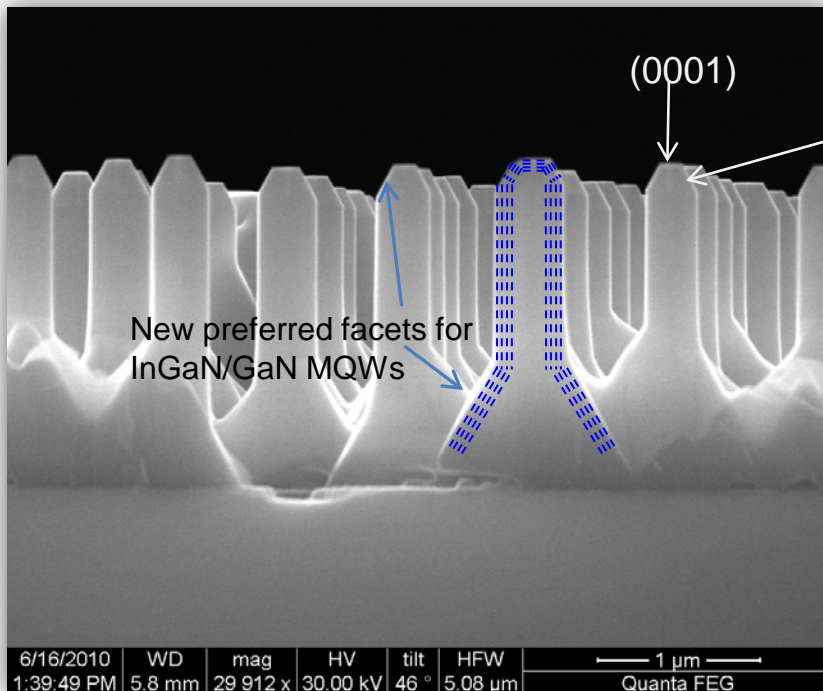
- nanowire LED: comparable IQE to planar LED but peak IQE occurs at much lower pumping power (enhanced light absorption, heating)

Q. Li et al., *Optics Express* **19**, 25528 (2011)

Radial core-shell InGaN/GaN MQWs on top-down NWs

- Radial core-shell heterostructures
- Much higher active area than axial or planar structures
 - Reduced strain InGaN growth for higher In incorporation

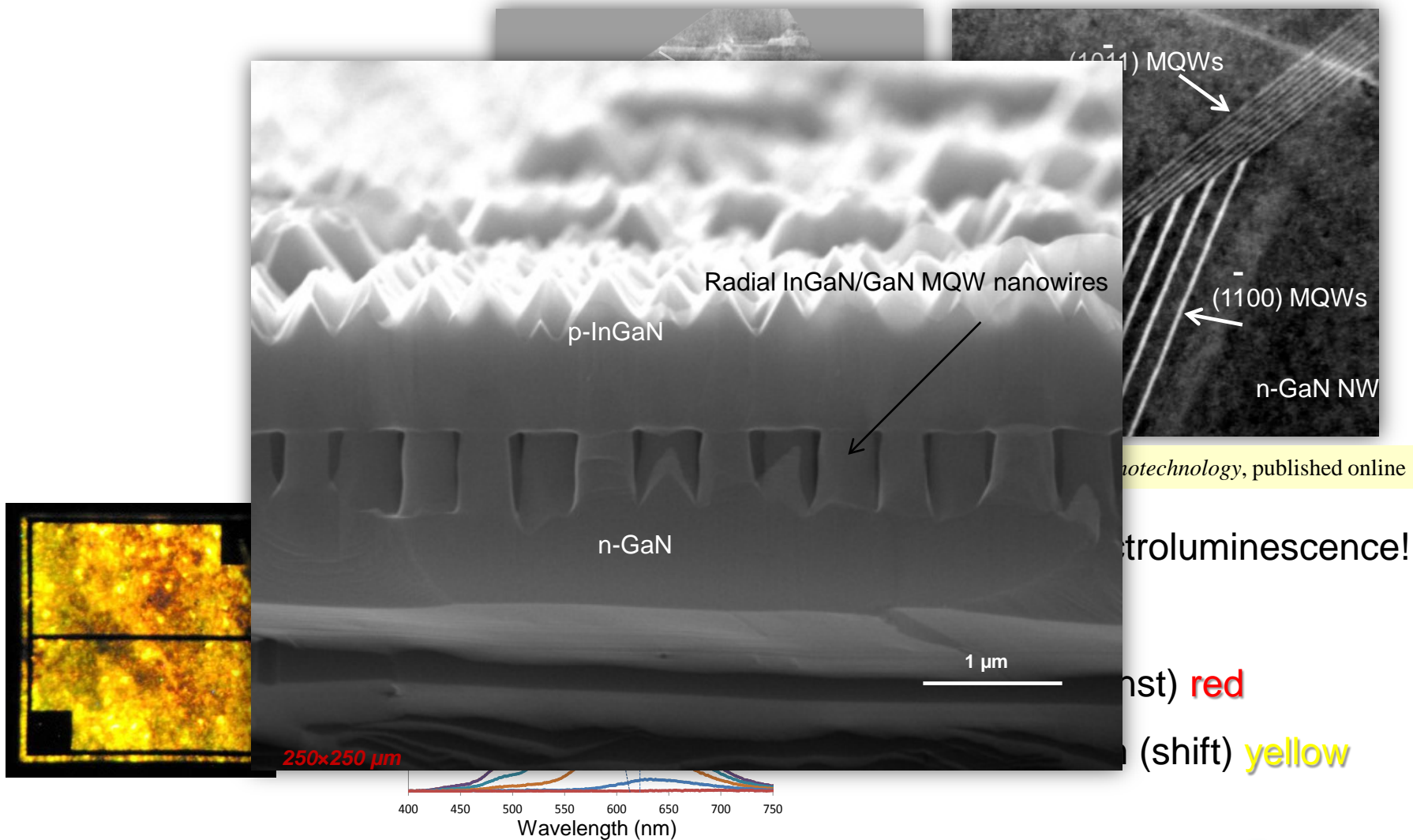
After 5-period MQW GaN/InGaN shell growth



New semipolar facets form with InGaN/GaN MQW growth

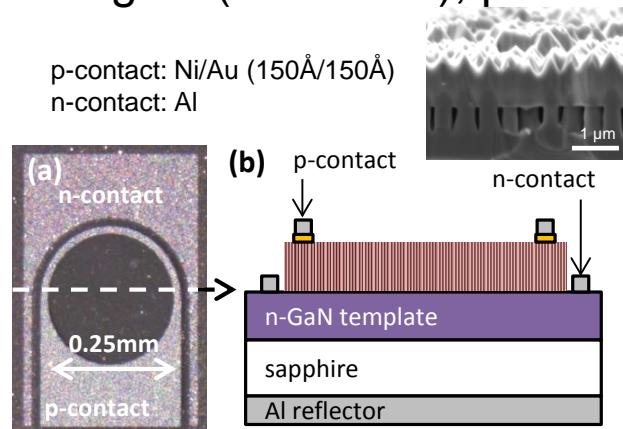


Highlight: Electrically injected core-shell nanowire based "3D" LED emitting at yellow-red wavelengths

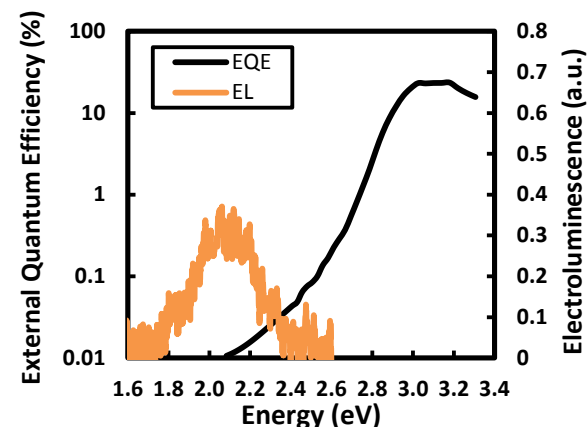
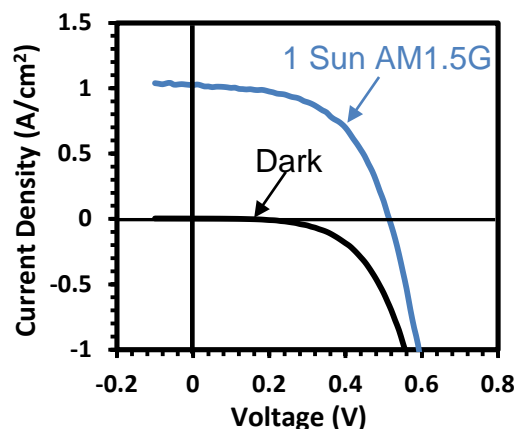


III-nitride nanowire arrayed solar cell

- III-nitride solar cells: InGaN bandgap (0.7-3.4 eV) covers solar spectrum; high rad. resistance
- Nanowire solar cells: increased light scattering/absorption, short carrier collection lengths (core-shell), potentially smaller bandgap cell (higher In content InGaN layers)



J. Wierer et al., *Nanotechnology*, in press



- First vertically integrated III-nitride nanowire solar cell*

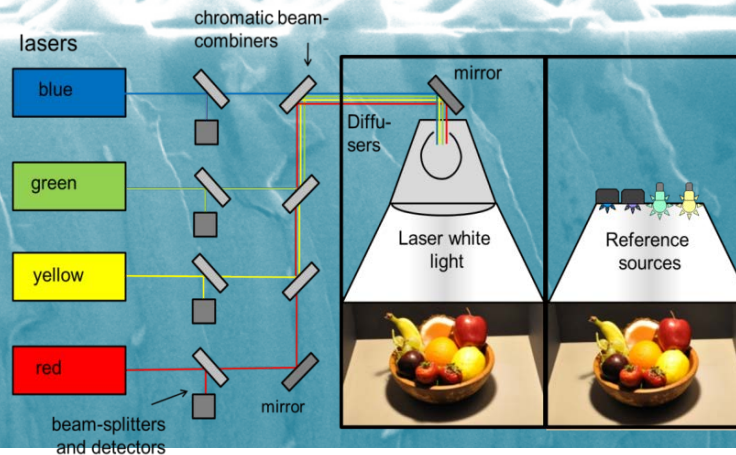
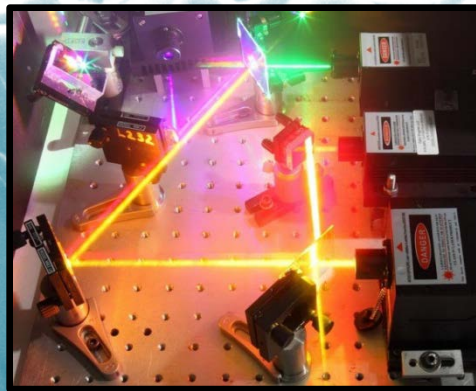
Previous Work: Single III-nitride NW solar cell:
Dong, Y. et al., *Nano Lett.* **9** 2183 (2009)

- Peak EQE ~23% at 3.0 eV; Photoresponse to 2.1 eV (590 nm), lowest bandgap reported for III-nitride solar cell; V_{OC} ~ 0.5 V; FF ~ 54%; Power conversion efficiency ~0.3% (shorting from defects in nanowire templates)

Nanowire Lasers

Why Nanowire Lasers?

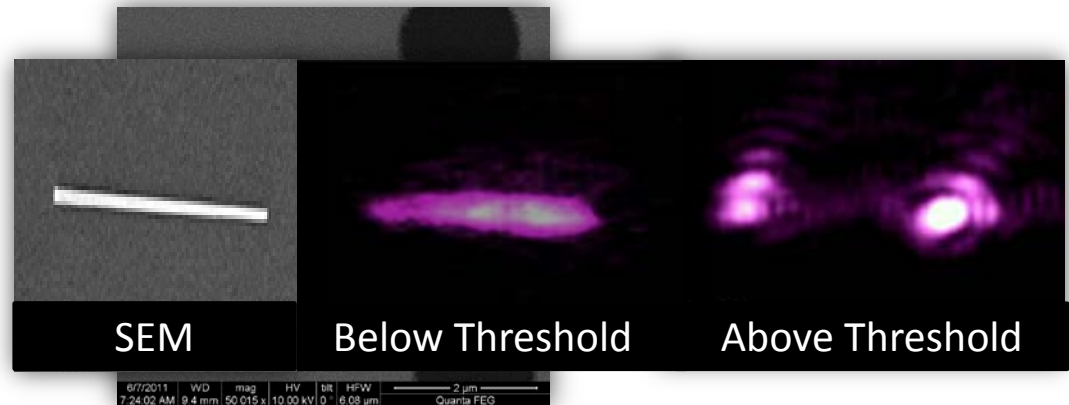
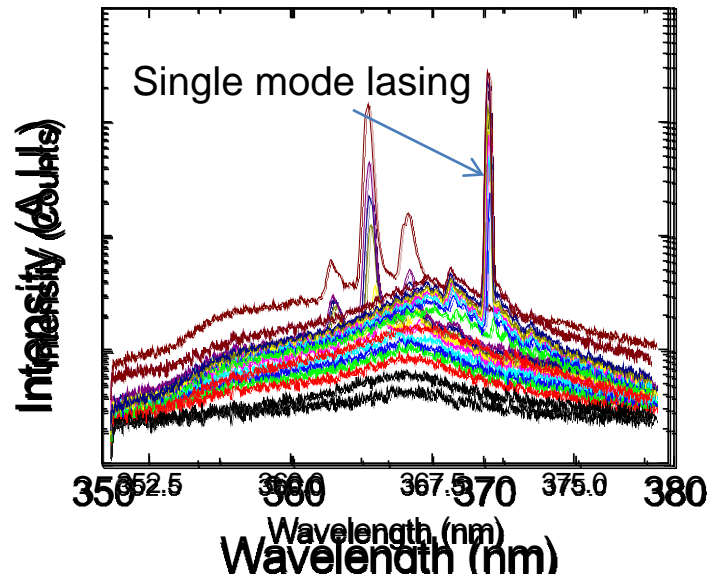
- *Nanowire forms a low loss optical cavity*
- *Low threshold and low power due to small mode volume*
- *Possibility of high efficiency lasers at green and yellow wavelengths*
- *Potential applications including electronic/optical integration, sensing, imaging, lithography, **lighting***
- *Lasers may circumvent droop problem in LEDs*



A. Neumann et al, *Optics Express*, **19**, A982-A990 (2011)

Single mode GaN nanowire laser

- Single-mode lasers are desirable (no mode-hopping, lower threshold & noise)



Nanowire dimensions: ~130nm x 4.7µm
Nanowire dimensions: ~500 nm x 4.7µm

Single-mode Linewidth (<0.1 nm), **High Power** (150 mW), **High Mode Suppression Ratio**, and **Low Threshold** (~250 kW/cm²)

Modeling shows that by reducing the dimensionality of the wire the number of competing modes is reduced, leading to single-mode lasing.



Summary

- **Nanowire based architectures have several potential advantages over planar-based SSL**
- **Numerous scientific & technical challenges for nanowire-based SSL**
- **Our efforts to address these challenges combine nanowire synthesis/fabrication, nanocharacterization, and devices**
- **Research highlights**
 - **Growth and properties of bottom-up III-nitride nanowires**
 - **New top-down nanowire fabrication for controlled geometries, lower point densities, vertical device integration**
 - **Top-down nanowire LEDs/solar cells/lasers**

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