



# Strength of Materials at High Pressures and Strain Rates

**Tracy Vogler**

**Department 8246, Mechanics of Materials  
Sandia National Laboratories, Livermore CA**

**Applied Physics Laboratory**

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# Acknowledgements

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- Staff - C.S. Alexander, T. Ao, J.-P. Davis, D. Dolan, M. Furnish, M. Knudson, W.D. Reinhart, S. Root, J. Wise
- Students - J. Brown, D.A. Fredenburg
- STAR - T. Thornhill, J. Martinez, R. Palomino, H. Anderson
- Z - the large crew for design, assembly and operations
- retired colleagues: Lalit Chhabildas, Dennis Grady, Jim Asay

*numerous technicians and support staff  
are essential to facility operations*



# Outline of Talk

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- **Introduction to shock and high-pressure physics**
- **Gas gun technology and examples**
- **The Z machine and examples**
- **Background on high pressure strength and techniques to measure it**
- **Strength of ceramics under shock loading**
- **Strength of metals under isentropic loading**
- **Conclusions**

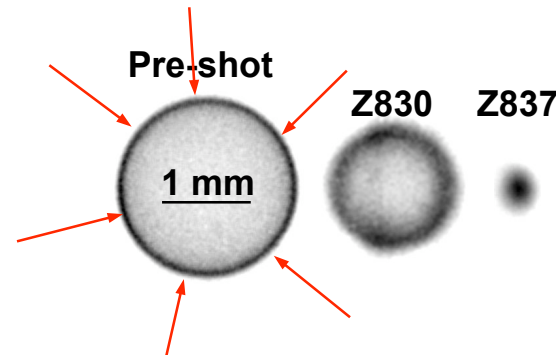


# Why Do We Need To Know the Behavior of Materials Under Extreme Conditions?

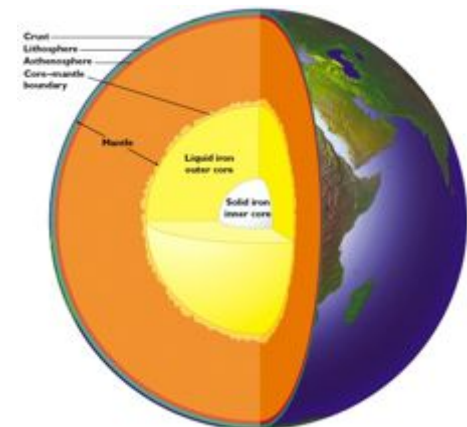
- weapons applications (warheads, armor, etc.)
- explosives behavior and applications



- inertial confinement fusion



- missile intercept; impact of asteroids or orbital debris ( $V=8-15$  km/s)
- planetary science ( $P \sim 360$  GPa,  $T \sim 7000$  K)

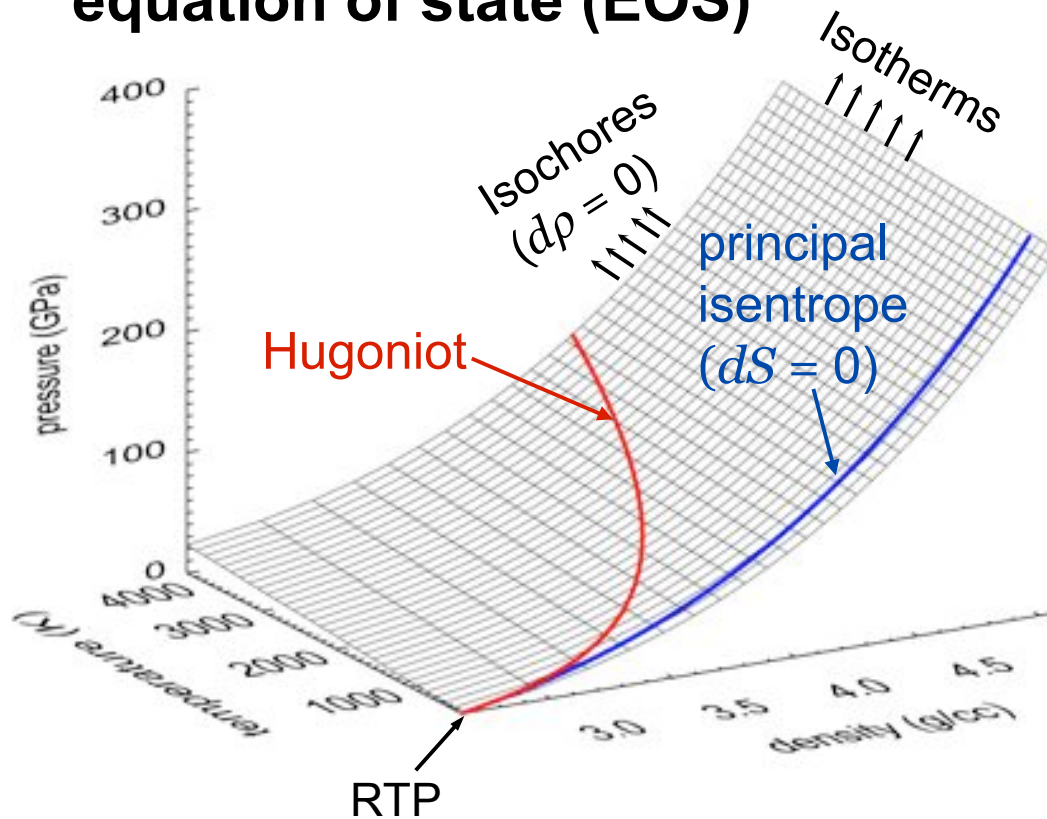




# Material Behavior: EOS & Constitutive Aspects

M. Knudson,  
M. Desjarlais

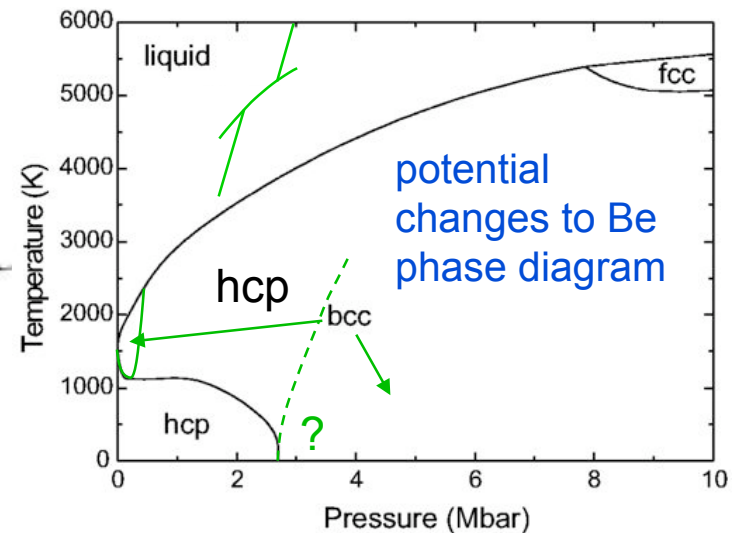
## equation of state (EOS)



one thermodynamic  
state variable as a  
function of two others:

pressure  $P = P(r, T)$

Helmholtz energy  $f = f(v, T)$

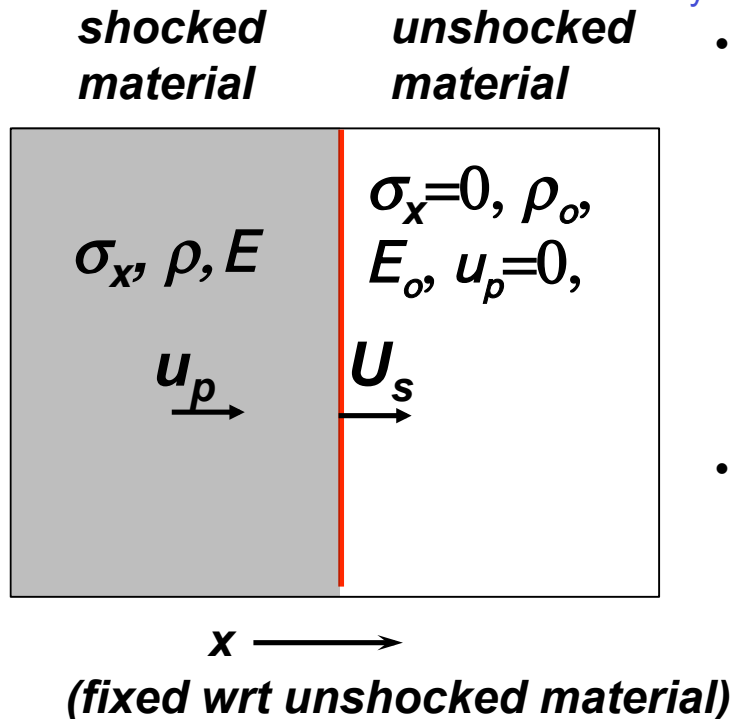


**Also: strength, damage, spall (tensile failure),  
compaction**



# What is a Shock Wave?

- A “discontinuous” wave that moves at a fixed velocity (if steady)
  - wave front moves at speed  $U_s$  (*shock velocity*)
  - shocked material moves at speed  $u_p$  (*particle or mass velocity*)
  - uniaxial strain condition ( $\epsilon_y = \epsilon_z = \epsilon_{xy} = \epsilon_{yz} = \epsilon_{xz} = 0$ )



- States ahead and behind shock assumed to be in thermodynamic equilibrium
  - well defined temperature in each state
  - described by equilibrium thermodynamics
- Shock compression is adiabatic
  - very fast process ( $< 1$  ns)
  - irreversible (i.e. NOT isentropic)
  - temperature *typically* increases



# Conservation Equations and the Shock Hugoniot

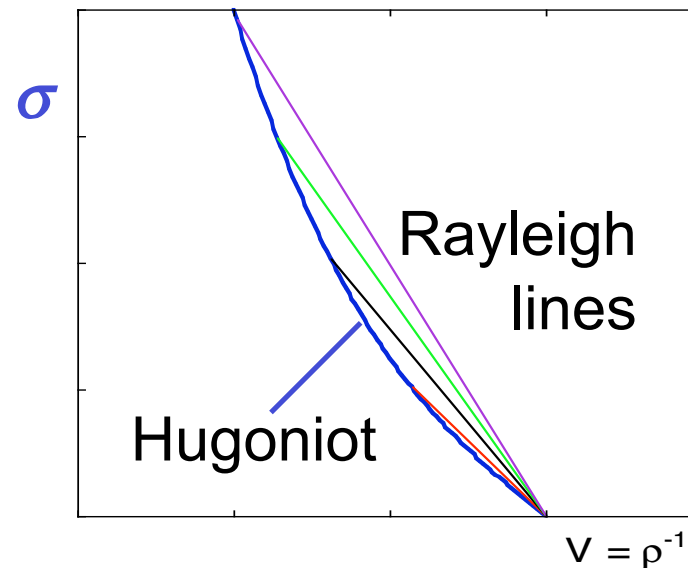
- Five variables:  $\sigma_x$ ,  $u_p$ ,  $U_s$ ,  $\rho$ , and  $E$
- Three conservation relationships (Rankine-Hugoniot jump conditions)
  - By measuring two variables (typically  $\sigma_x$ ,  $u_p$ , or  $U_s$ ), the other three can be determined

## conservation of

mass:  $\rho_o U_s = \rho (U_s - u_p)$

momentum:  $\sigma_x = \rho_o U_s u_p$

energy:  $E - E_o = 0.5 \sigma_x (V_o - V)$



material loads along the Rayleigh line, so the Hugoniot is a collection of end states, not a material response curve  
*the Hugoniot is not a complete equation of state (EOS)!*



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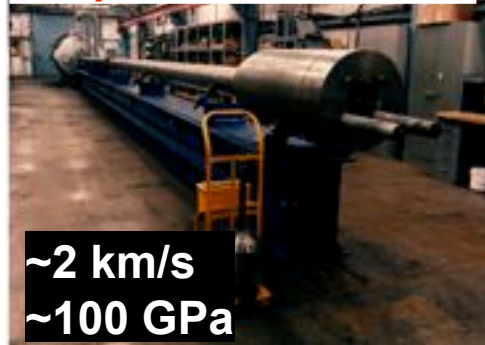
# Gas Guns to Generate Shock Waves



**Single Stage Gun 100mm**



**Propellant Gun 89mm**



**Two-Stage Gun 29mm**



## gas guns

- launch thin plates (mm's) at high velocities
- well-posed, repeatable initial conditions
- sample is in uniaxial **strain**
- used to study material behavior at high pressures and strain rates
- usable in laboratory setting

**Three-Stage Gun 17mm**



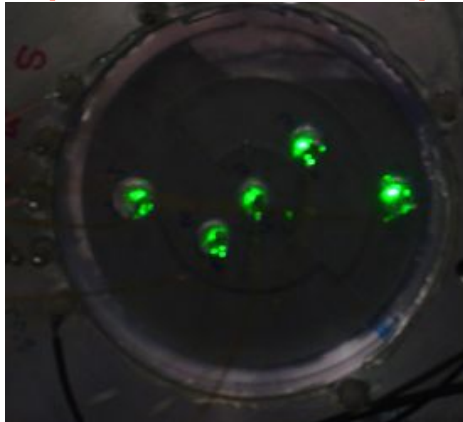
Chhabildas, L. C., Dunn, J. E., Reinhart, W. D., and Miller, J. M. (1993). "An impact technique to accelerate flier plates to velocities over 12 km/s," *Int. J. Impact Eng.* **14**, 121-132.

*also: explosives, lasers, magnetic loading (Z)*

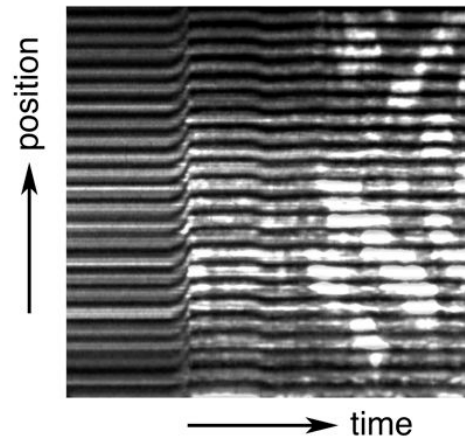


# Diagnostics for Dynamic Experiments

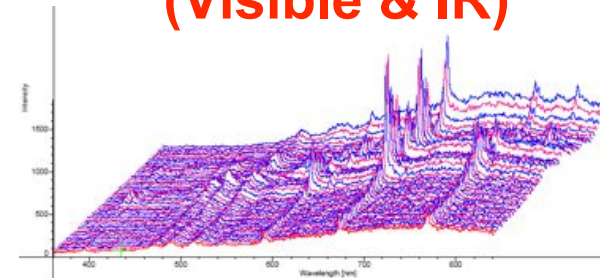
## Velocity Interferometry (VISAR & PDV)



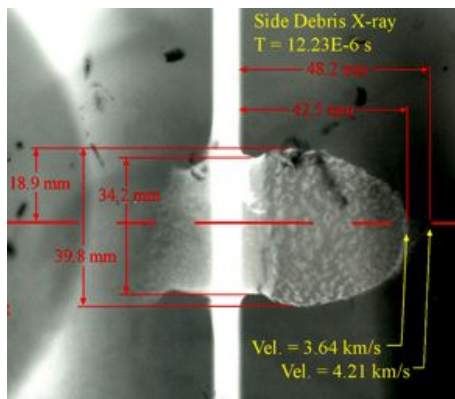
## Line-VISAR



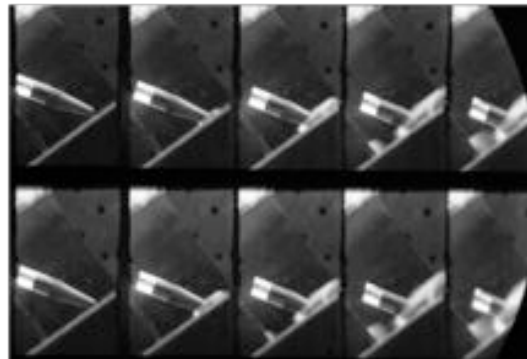
## Time-Resolved Spectroscopy (Visible & IR)



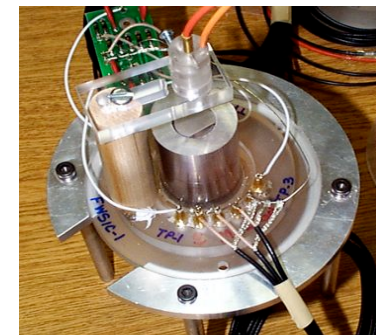
## Flash X-rays



## High-Speed Photography



## Pressure Gauges



Advanced Diagnostics: pRad, synchrotron, etc.

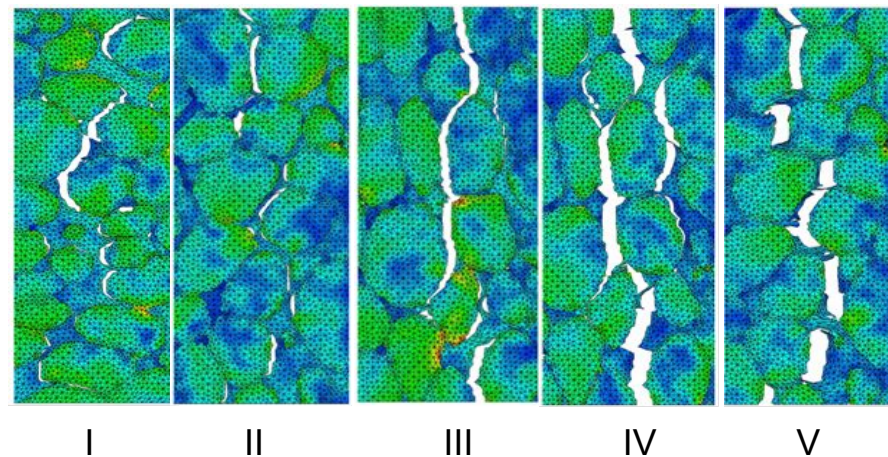
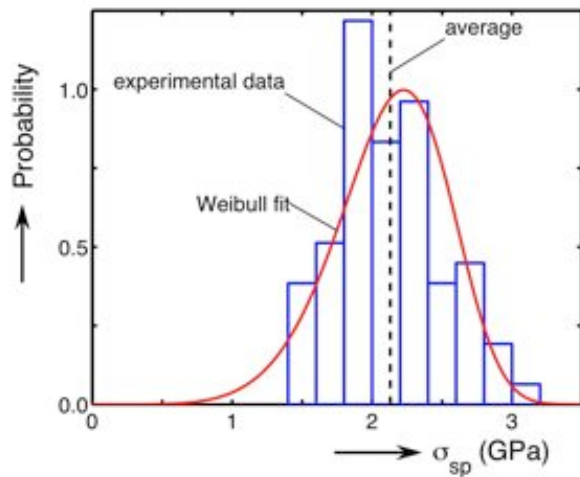
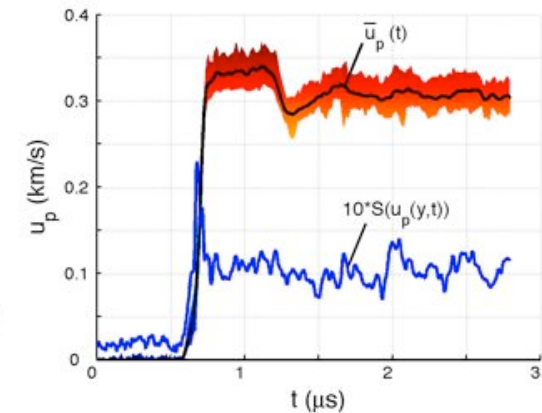
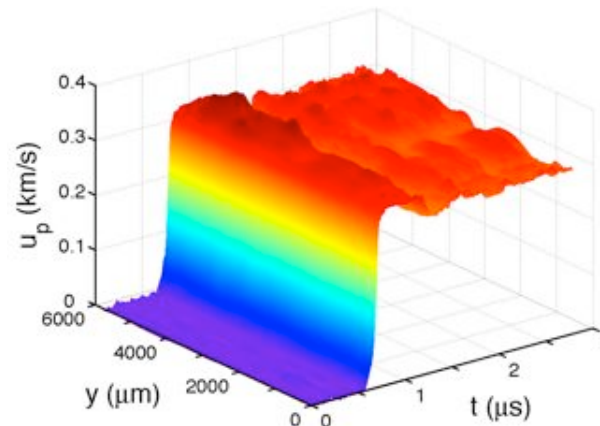
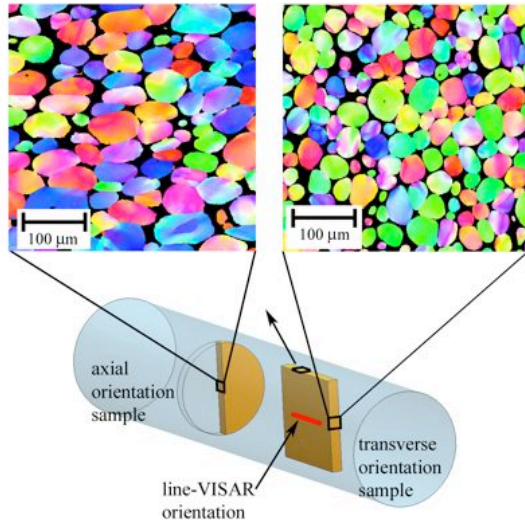


# Heterogeneous and Statistical Aspects of Spall

J. Clayton,  
T. Vogler



WHA - tungsten heavy alloy

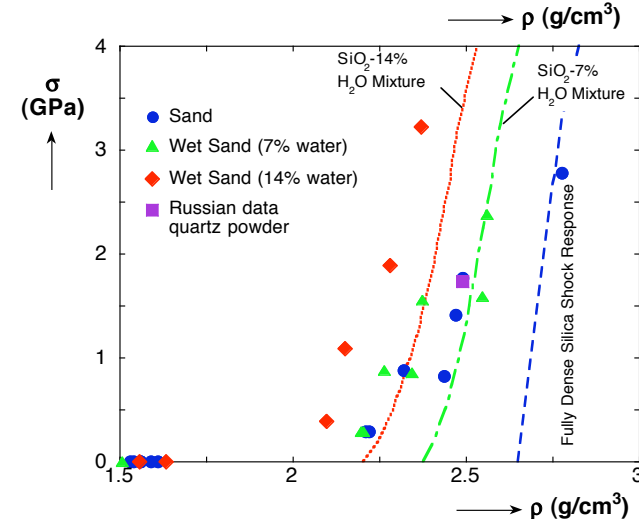
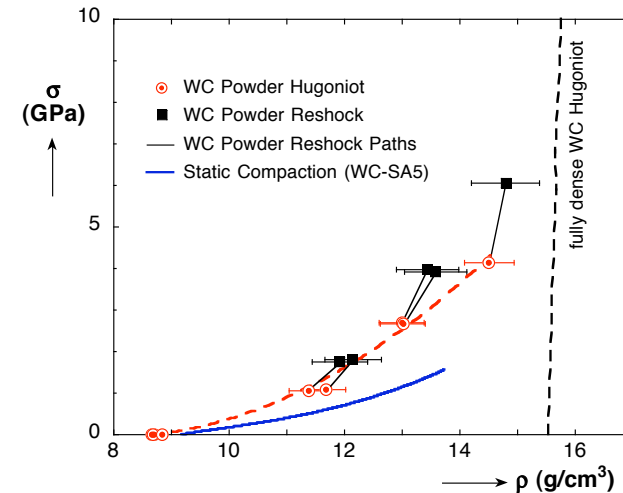
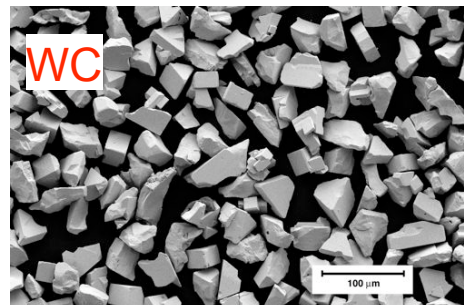
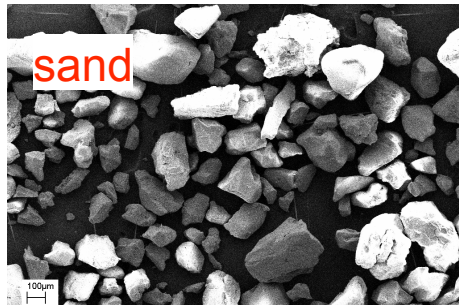
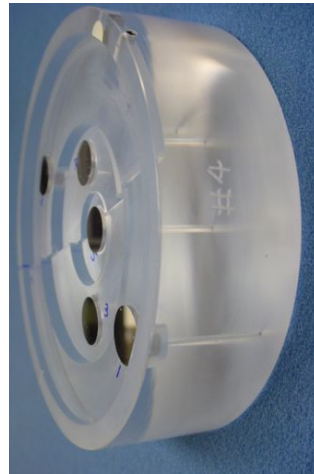
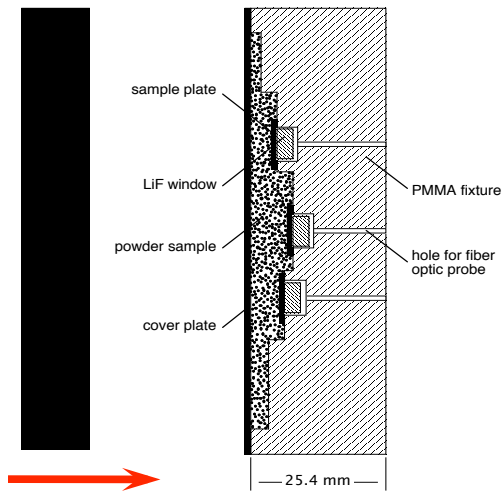


Vogler, T. J., and Clayton, J. D. (2008). "Heterogeneous deformation and spall of an extruded tungsten alloy: plate impact experiments and crystal plasticity modeling," *J. Mech. Phys. Solids* 56, 297-335.



# Compaction of Granular Ceramics

D. Grady,  
T. Vogler



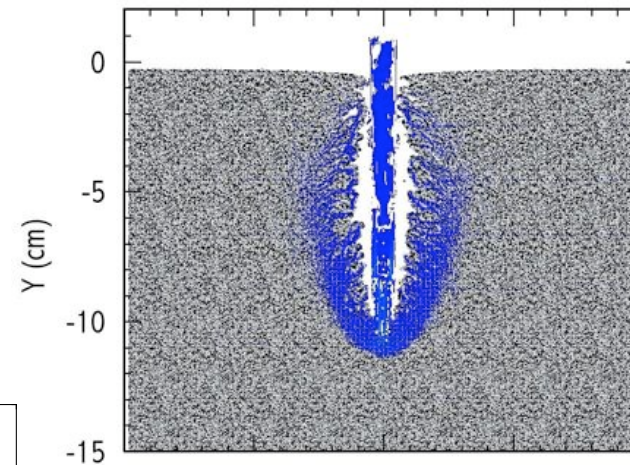
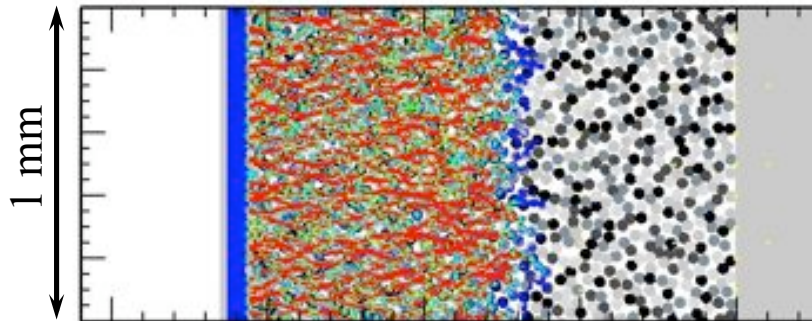
Vogler, T.J., Lee, M.Y., Grady, D.E., 2007. "Static and dynamic compaction of ceramic powders." *International Journal of Solids and Structures* **44**, 636-658.

Brown, J.L., Thornhill, T.F., Reinhart, W.D., Chhabildas, L.C., Vogler, T.J., 2007. "Shock response of dry sand." in *Shock Compression of Condensed Matter – 2007*, American Institute of Physics, 1363-1366.

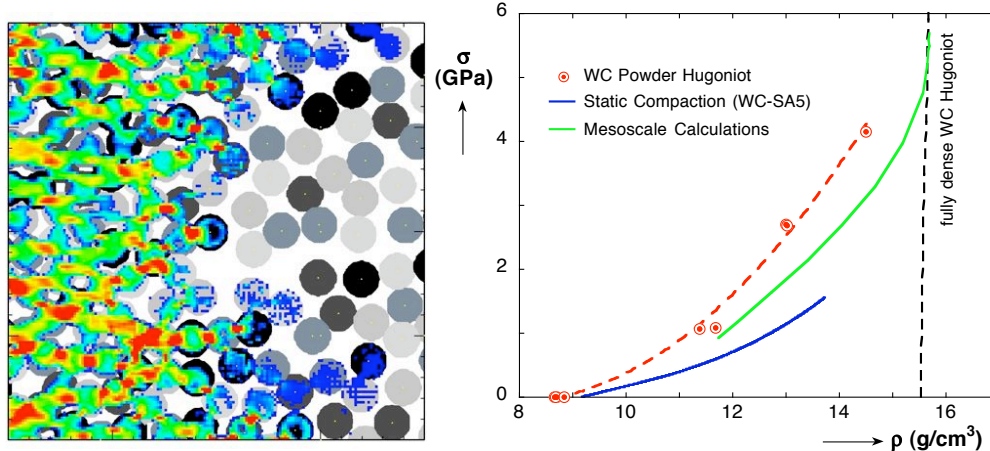


# Mesoscale Modeling of Granular Materials

J. Borg,  
T. Vogler

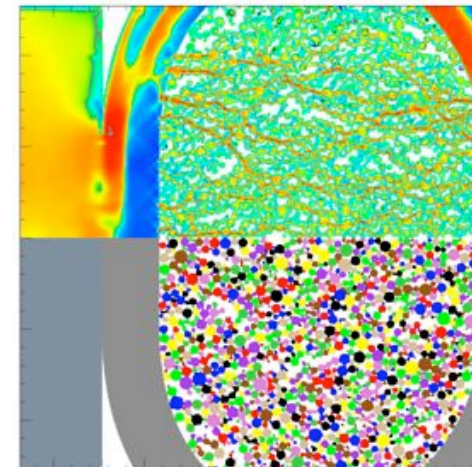


Borg, J. P., and Vogler, T. J. (2008). "Mesoscale simulations of a dart penetrating sand," *Int. J. Impact Eng.* **35**, 1435-1440.



Borg, J. P., and Vogler, T. J. (2008). "Mesoscale calculations of the dynamic behavior of a granular ceramic," *Int. J. Solids Structures* **45**, 1676-1696.

Borg, J. P., and Vogler, T. J. (2009). "Aspects of simulating the dynamic compaction of a granular ceramic," *Modelling Simul. Mater. Sci. Eng.* **17**, 045003.



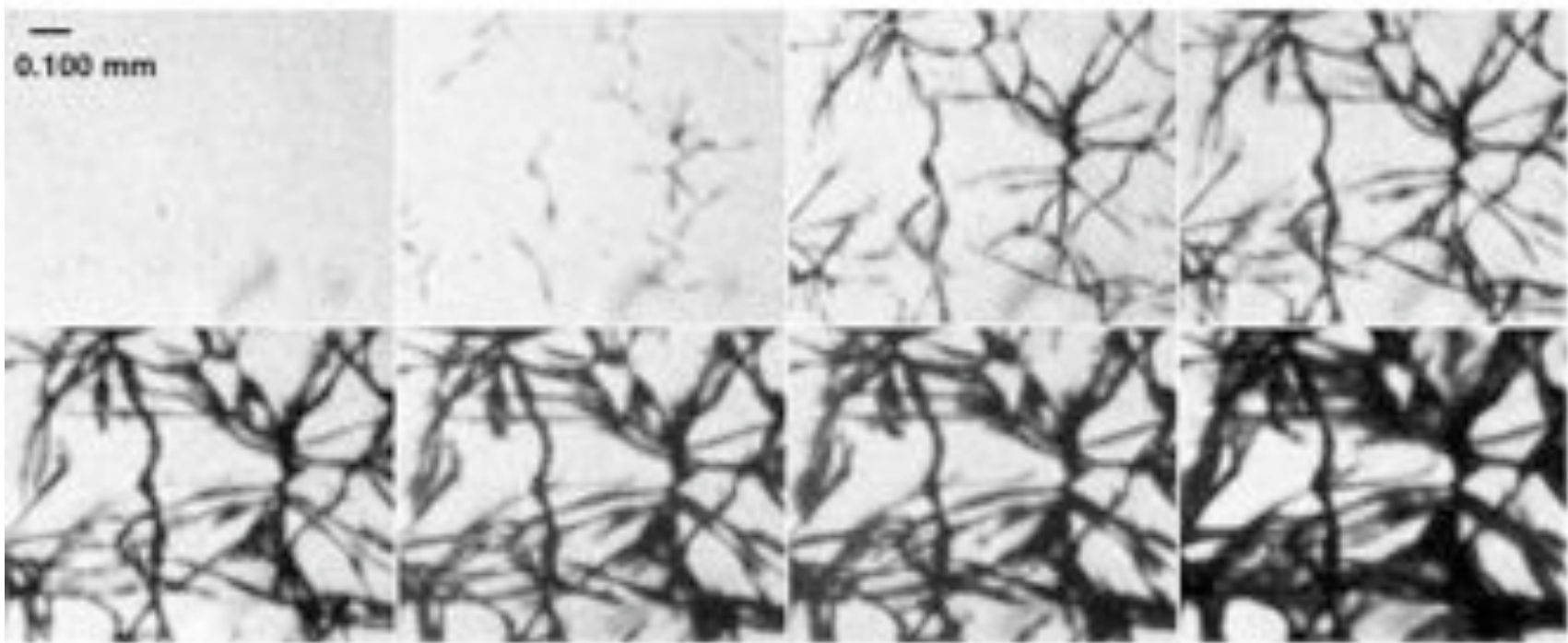


# Nanosecond Freezing of Water

- 2.7 GPa (ice VII)
- 25 ns exposure

t(ns) from  
shock arrival=

230	330	530	630
730	830	930	1530



D.H. Dolan and Y.M. Gupta, J. Chem. Phys. **121**, 9050 (2004).



# Outline of Talk

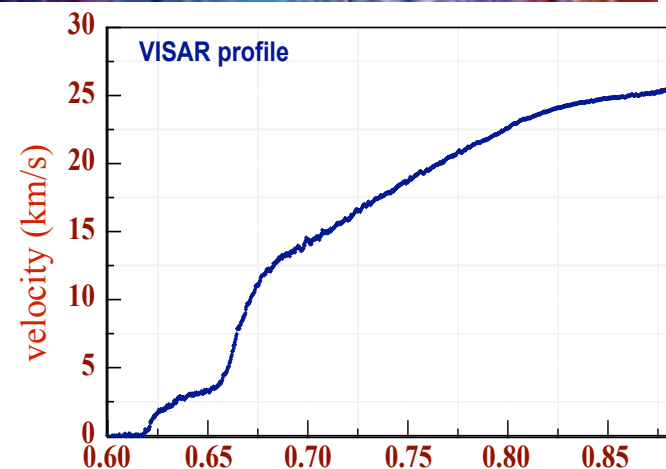
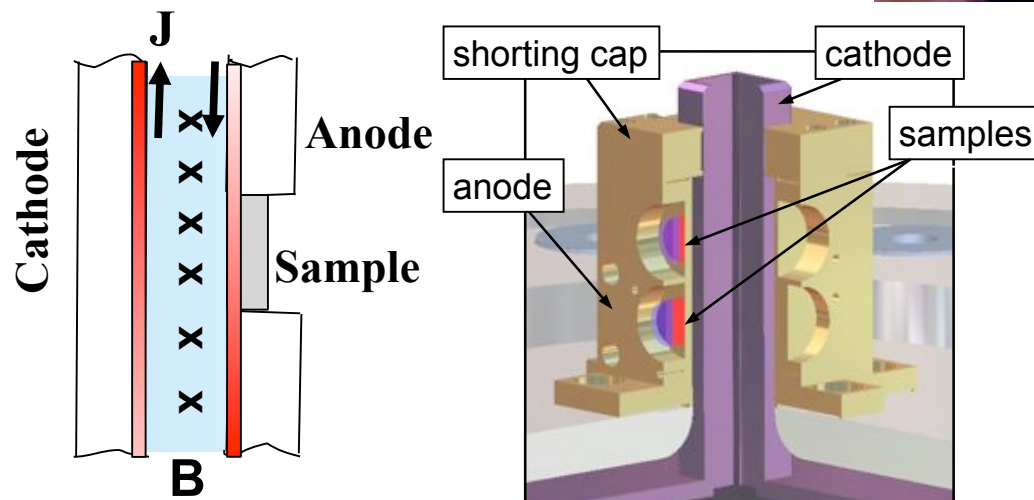
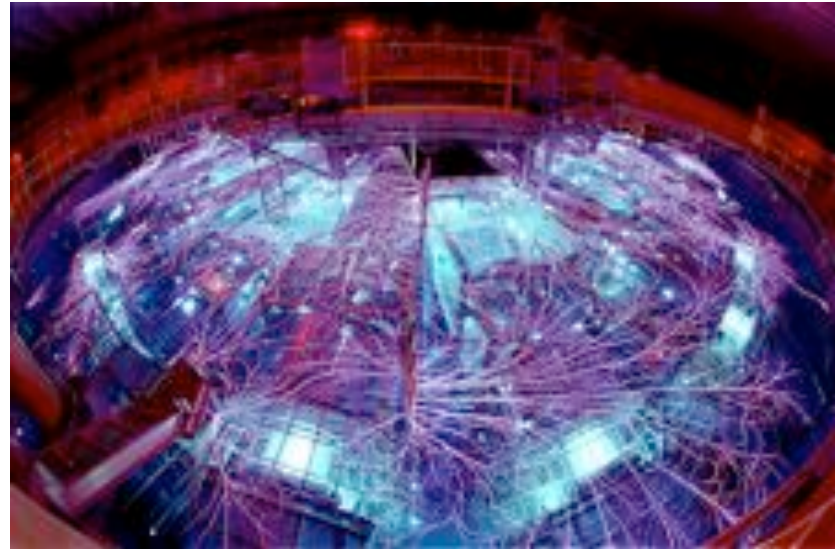
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# Z Pulsed Power Machine

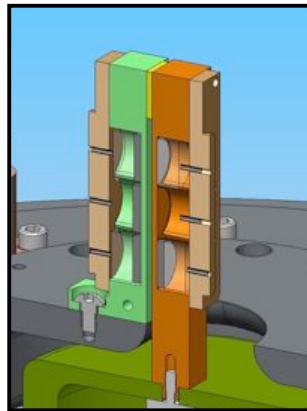
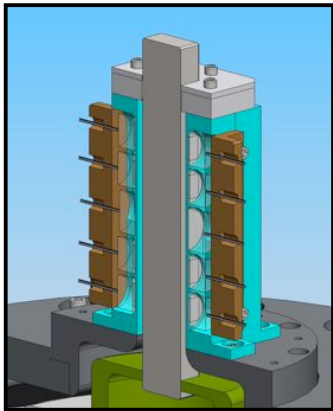
- Designed for ICF applications
- Generates ~26 MA over 100's of ns
- Utilize current to generate magnetic forces
- Magnetic forces create smooth waves in materials
- Waves used for isentropic loading (to ~400 GPa) and to launch high-velocity flyer plates (to ~40 km/s)



Davis, J.-P., Deeney, C., Knudson, M. D., Lemke, R. L., Pointon, T. D., and Bliss, D. E. (2005). "Magnetically driven isentropic compression to multimegabar pressures using shaped current pulses on the Z accelerator," *Physics of Plasmas* 12, 056310.



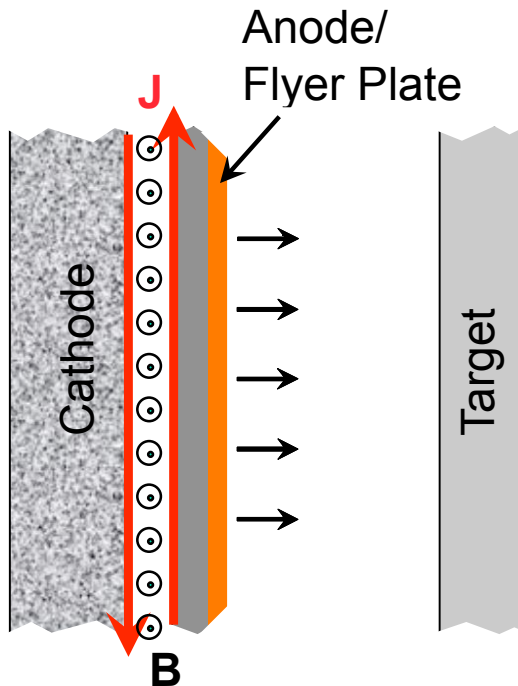
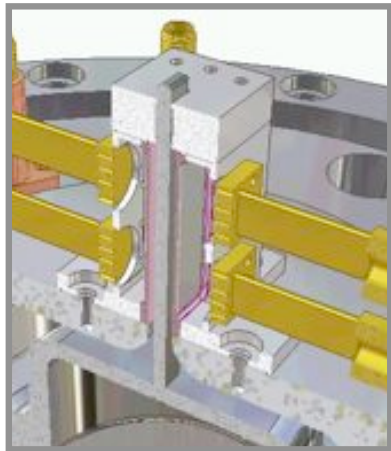
# Two Approaches Developed for Accurate EOS Studies – Both Major Advances



## Isentropic Compression Experiments (ICE)\*

Magnetically driven Isentropic Compression Experiments (ICE) to provide measurement of continuous compression curves to ~4 Mbar  
- previously unavailable at Mbar pressures

\* Developed with LLNL



## Magnetically launched flyer plates

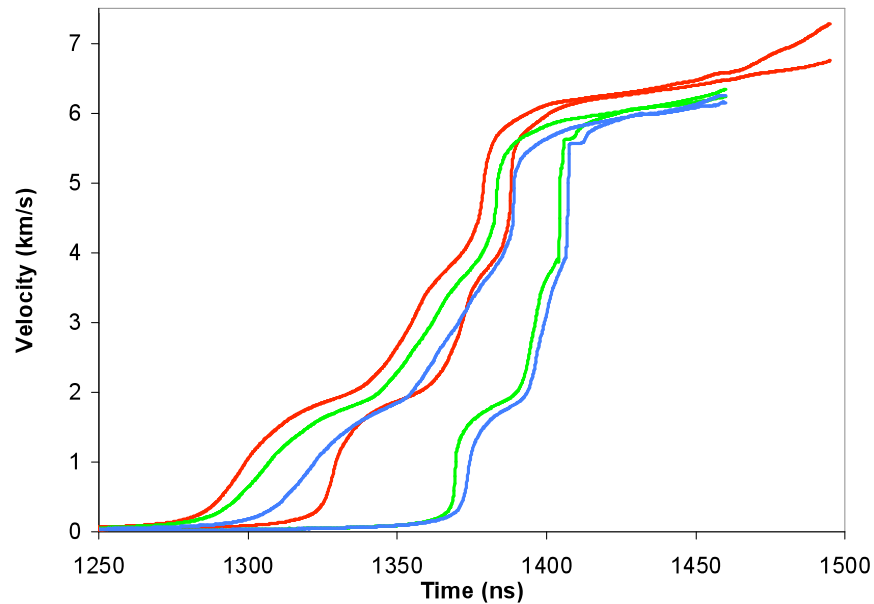
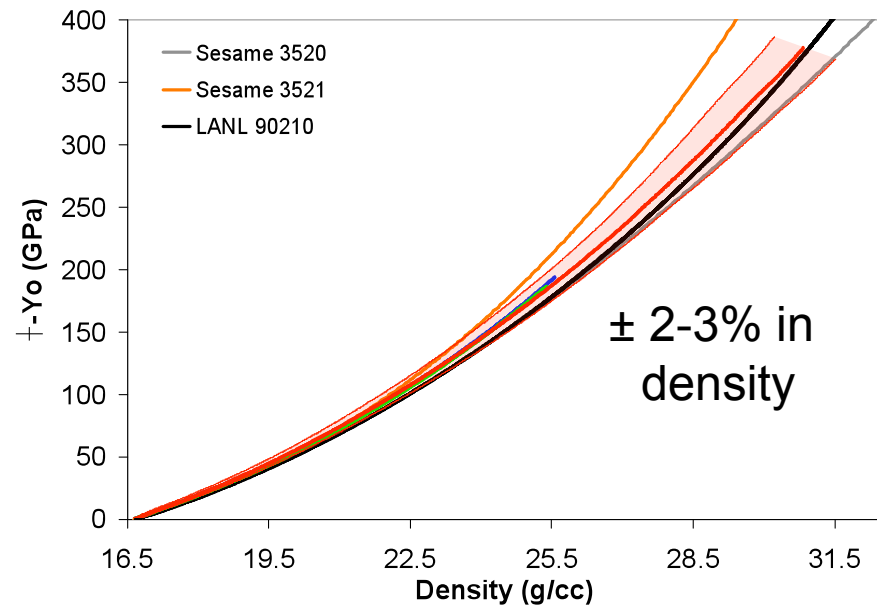
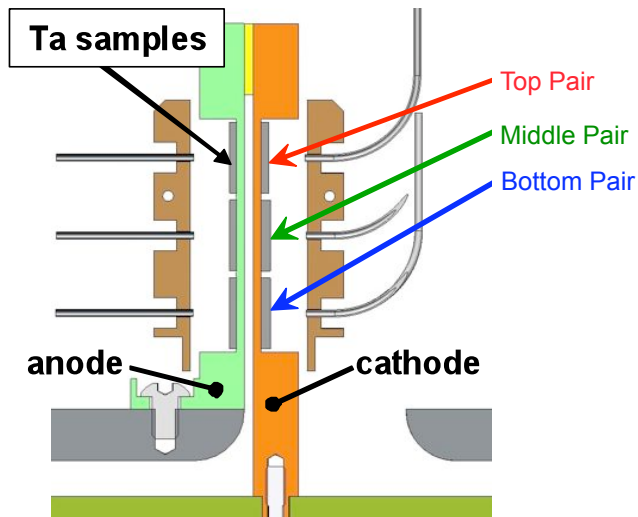
Magnetically driven flyer plates for shock experiments at velocities to > 40 km/s  
exceeds gas gun velocities by > 5X  
and pressures by > 10X with  
comparable accuracy

Lemke, et al., J. Appl. Phys. 98, 073530 (2005)



# Quasi-Isentrope of Ta to nearly 400 GPa

J.-P. Davis

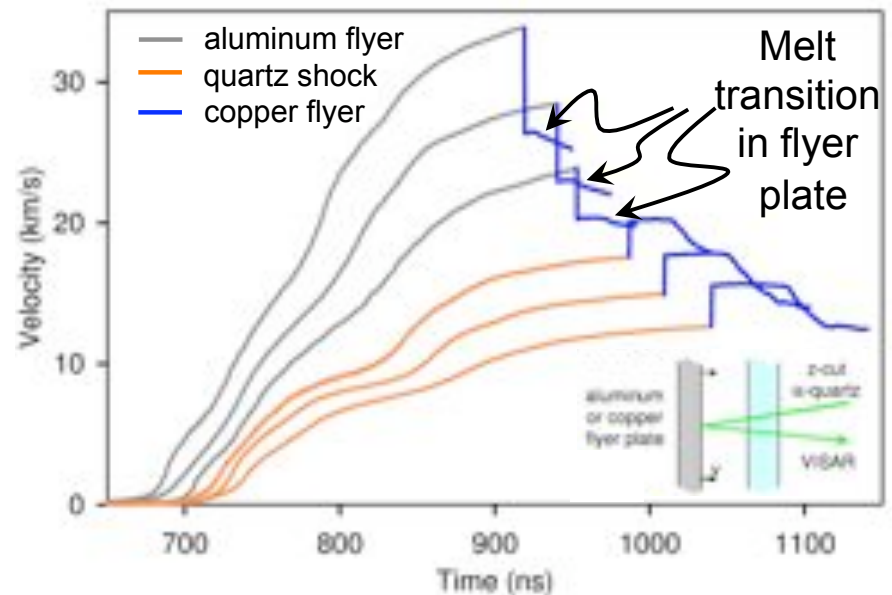
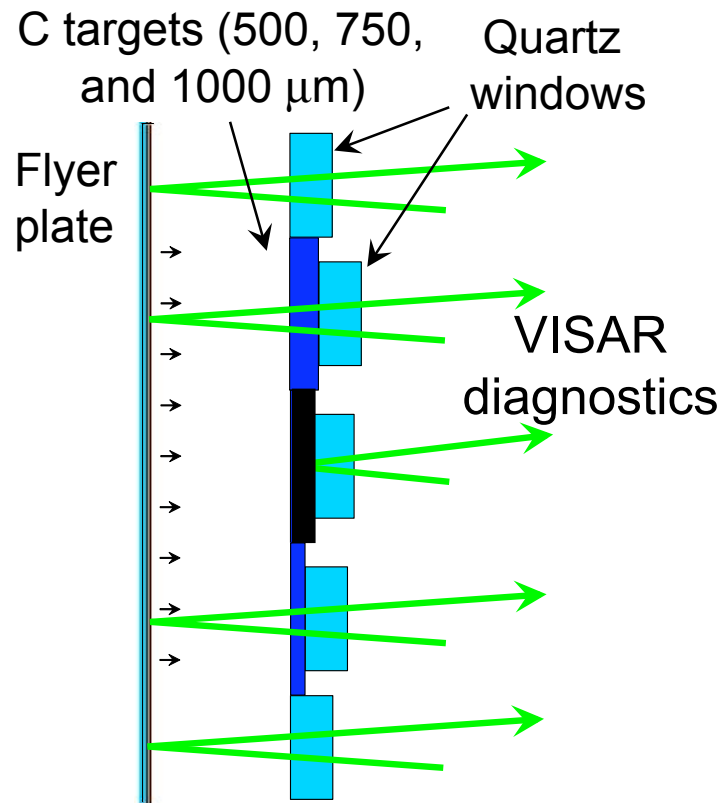


The extracted isentrope discriminates between various tabular equations of state for Ta



# Quartz of Interest as Standard - Window in Z Shots

M. Knudson  
M. Desjarlais



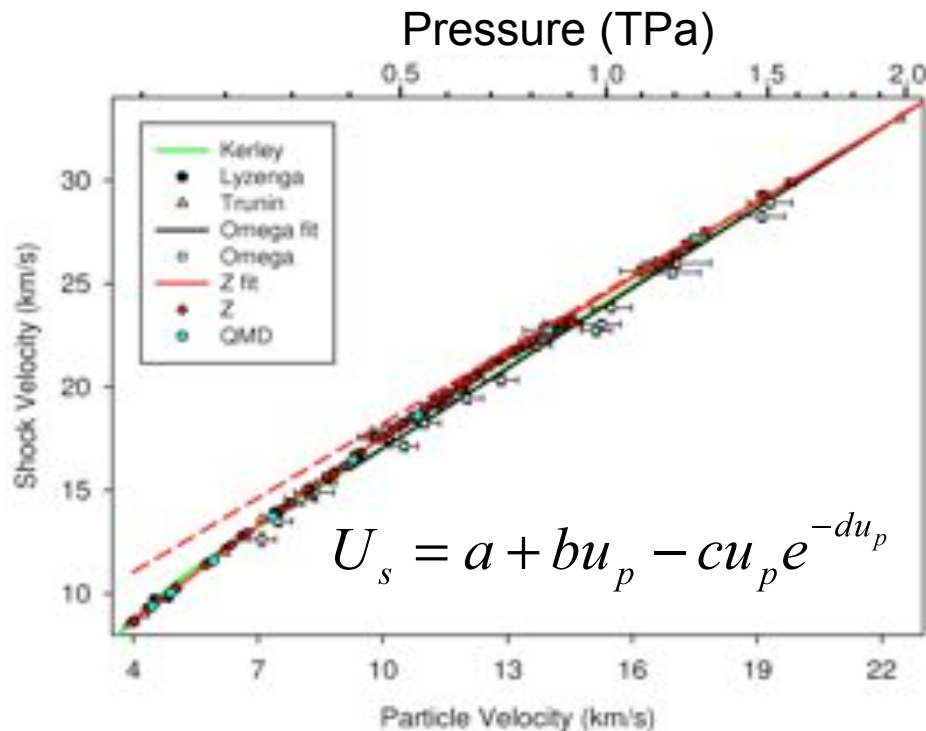
Knudson, M. D., M. P. Desjarlais and D. H. Dolan (2008). "Shock-wave exploration of the high-pressure phases of carbon." *Science* **322**: 1822-1825.

quartz data has large uncertainty and scatter, and  $U_s-u_p$  exhibits significant curvature in the several 100 GPa regime; attributed to dissociation

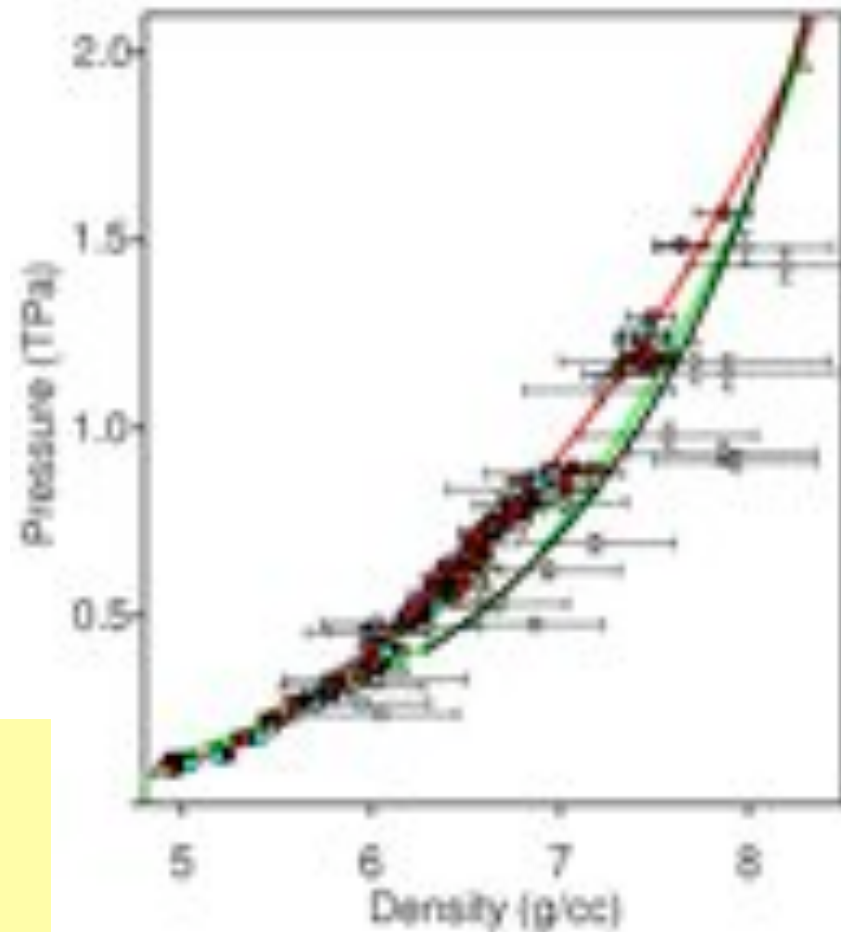


## New Hugoniot Data for $\alpha$ -Quartz to 1.5 TPa

M. Knudson  
M. Desjarlais



Data set includes ~150 points

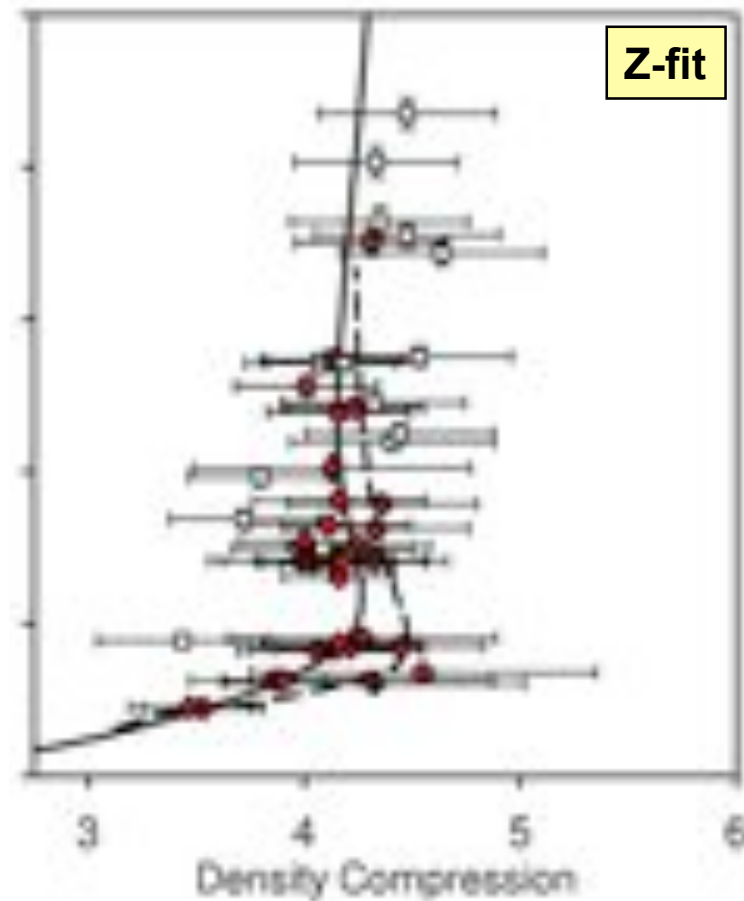
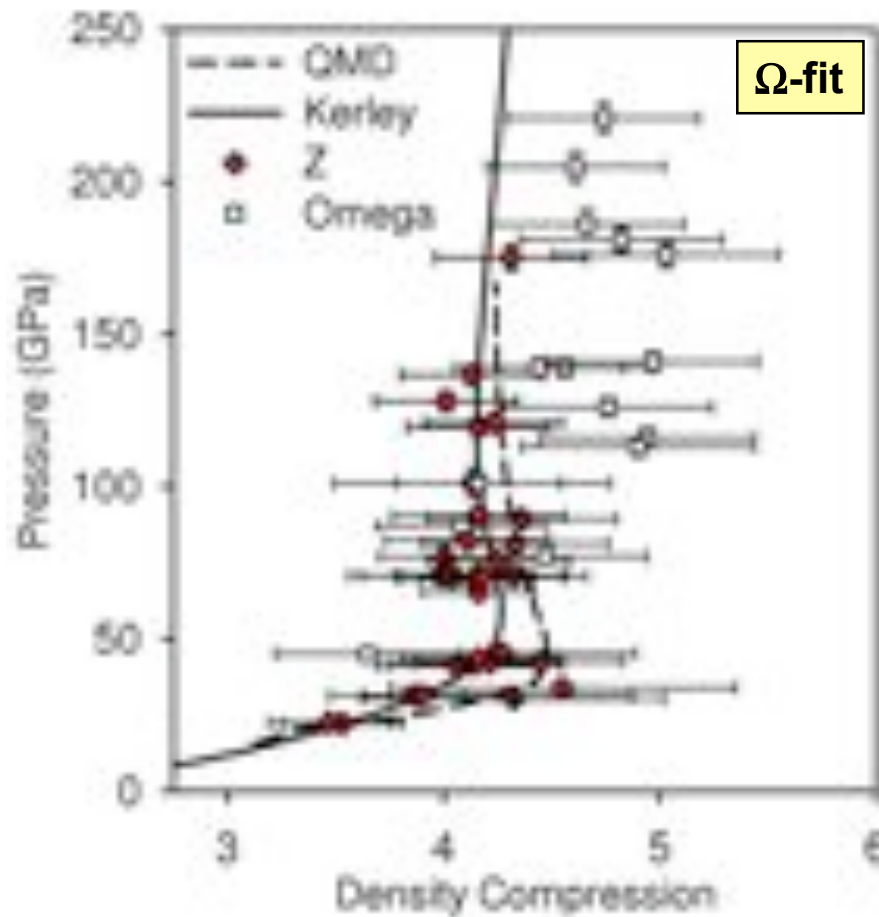


Knudson, M. D. and M. P. Desjarlais  
(2009). "Shock compression of quartz to  
1.6 TPa: redefining a pressure standard."  
*Physical Review Letters* **103**: 225501.



# Deuterium Results Affected by New Quartz Fit

M. Knudson  
M. Desjarlais



M.D. Knudson, et al., Phys. Rev. Lett. **87**, 225501 (2001); Phys. Rev. Lett. **90**, 035505 (2003); Hicks, et al., PRB 79, 014112 (2009)

Knudson & Desjarlais, PRL 103, 225501 (2009)



# Outline of Talk

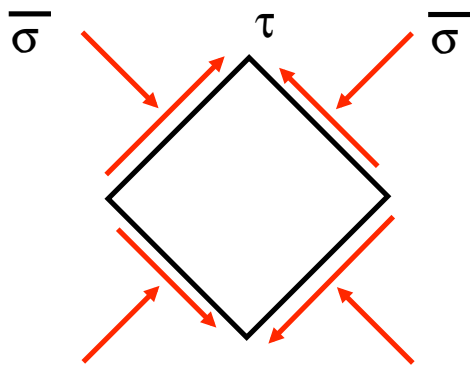
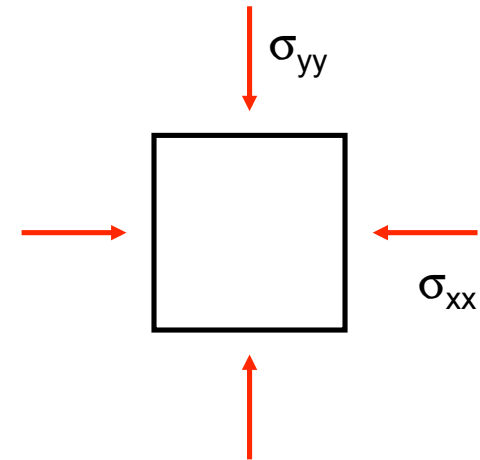
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# What is Strength?

- strength is the ability of a material to sustain  $\tau_{ij} \neq 0$  or  $\sigma_{xx} \neq \sigma_{yy}$
- for a 1-D shock or isentropic experiment,  $\sigma_{yy} = \sigma_{zz} \neq \sigma_{xx}$
- conservation equations provide no information about  $\sigma_{yy}$



By a simple tensor transformation (45° rotation), the stress state can be expressed as a mean stress (mechanical pressure) and a shear stress

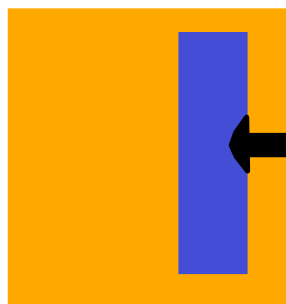
By analogy with uniaxial tension,  
 $Y = 2\tau_c$



# Why Do We Care About Strength?

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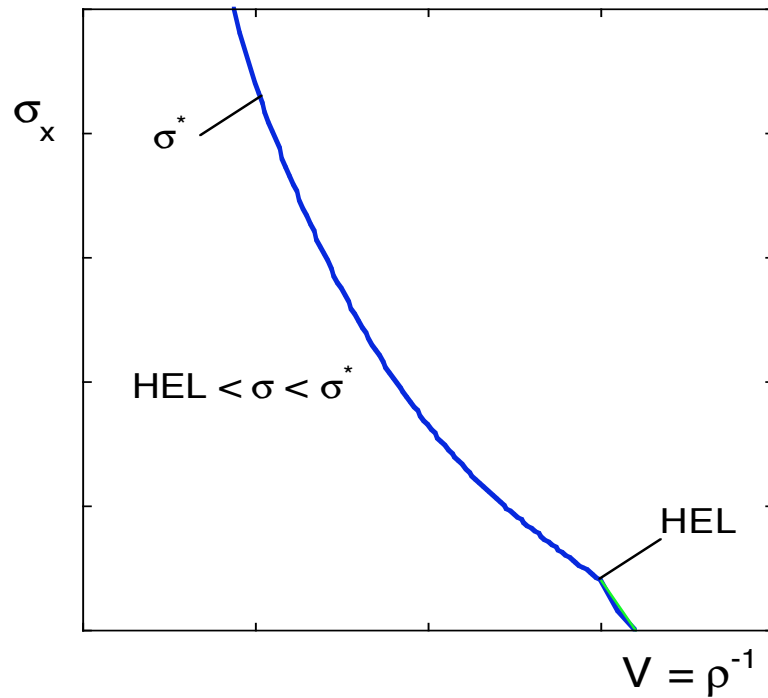
- increases uncertainties in EOS models
- strength affects stress state in diamond anvil cell
- understanding of strength needed for accurate computational results
- Rayleigh-Taylor instabilities inhibited by strength
- weapons and armor applications (ceramic armor, penetrators, etc.) influenced by strength



dwelt of penetrator  
on confined  
ceramic



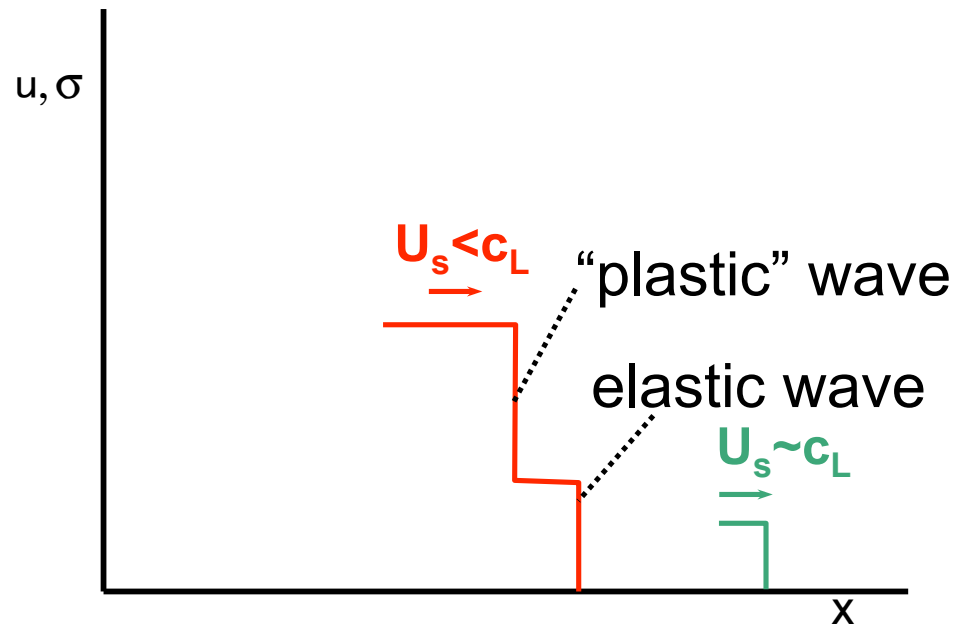
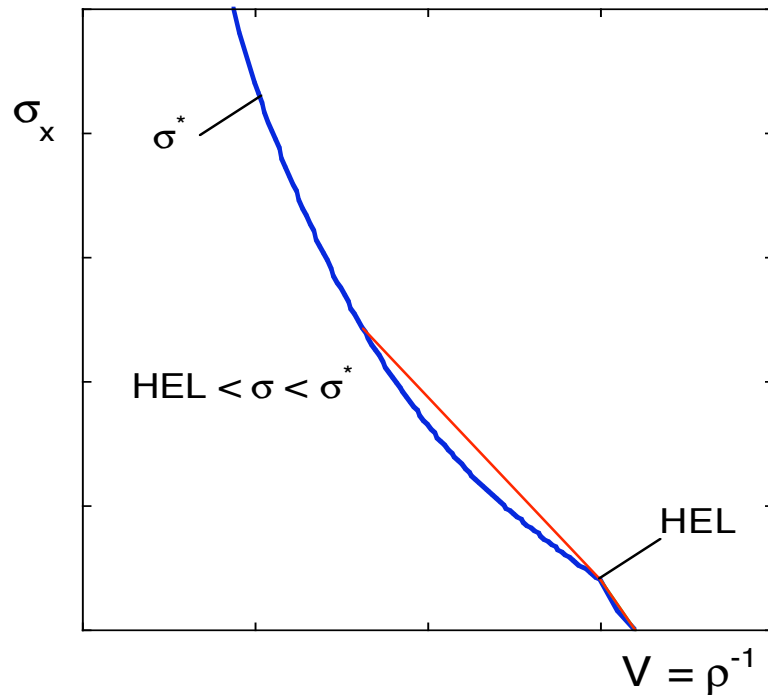
# Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses



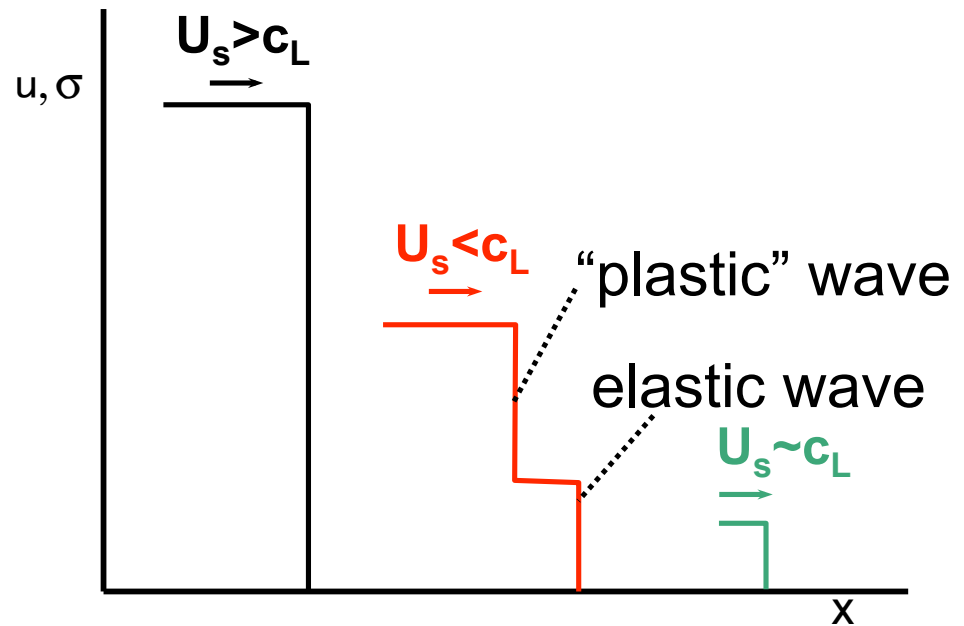
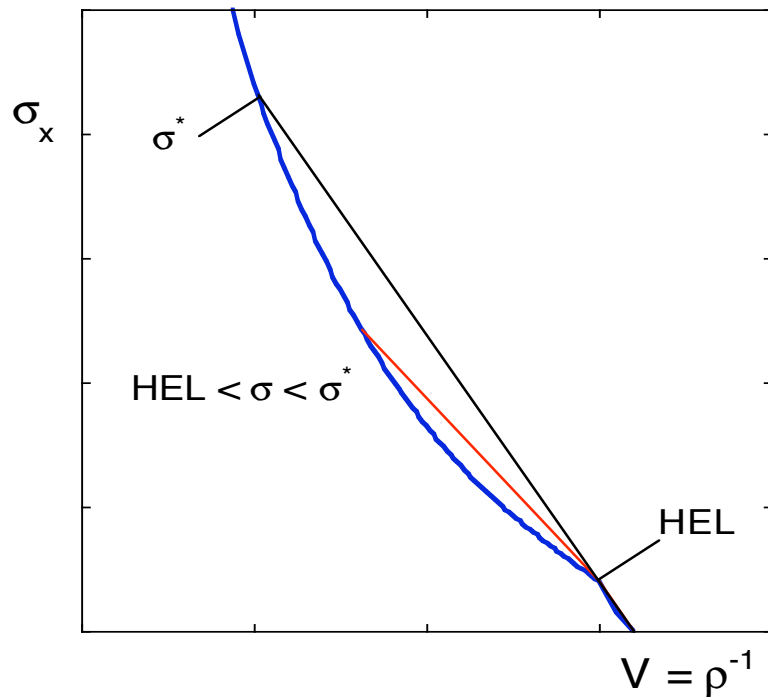
# Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses
- for stresses somewhat above HEL (Hugoniot elastic limit), a two-wave structure develops



# Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses
- for stresses somewhat above HEL (Hugoniot elastic limit), a two-wave structure develops
- for high stresses, plastic wave travels faster than elastic precursor

*above HEL,  $\sigma_y$  is unknown regardless of stress level!*



## Typical Values of HEL

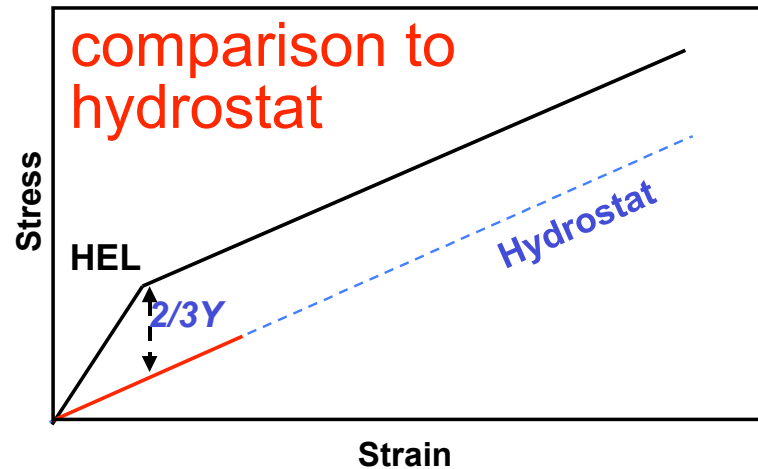
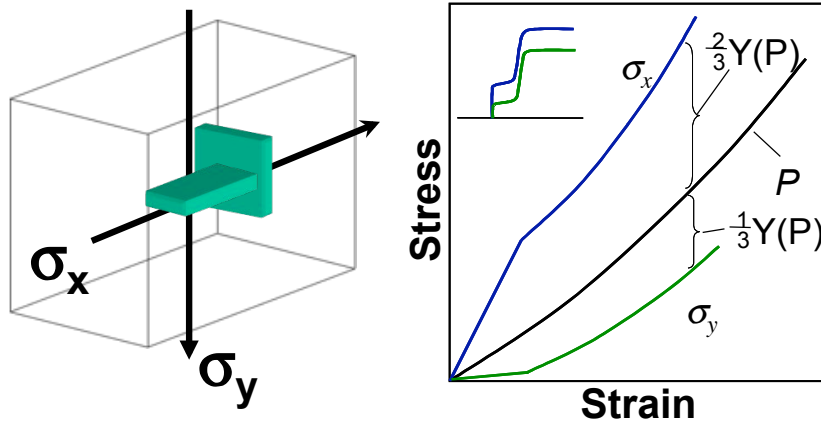
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Material	HEL (GPa)
6061-T6 Al	0.5
tungsten	3 - 4.5
silicon carbide (SiC)	12 - 14
boron carbide (B <sub>4</sub> C)	15 - 18
diamond	60 - 100



# Techniques to Determine Strength at High Pressures

## lateral gauges



- *stress gauges can provide independent measures of  $\sigma_x$  and  $\sigma_y$  to determine dynamic strength*
- *calibration of gauges difficult*
- *only function to ~20 GPa due to shorting of insulation*

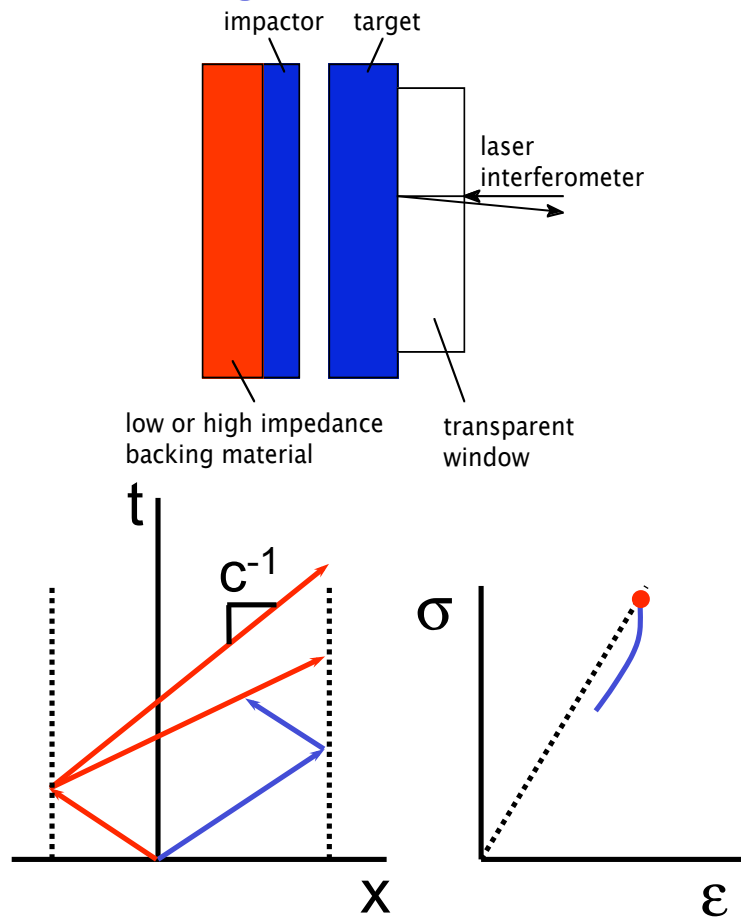
- *most common method: compare with hydrostatic data*
- *hydrostat from ultrasonic sound speed data ( $C$  vs.  $P$ ) or diamond anvil cell ( $P - V$ )*
- *uncertainties can be very large*

*also: X-ray diffraction, pressure-shear loading, growth of Rayleigh-Taylor instabilities, within diamond anvil cells*

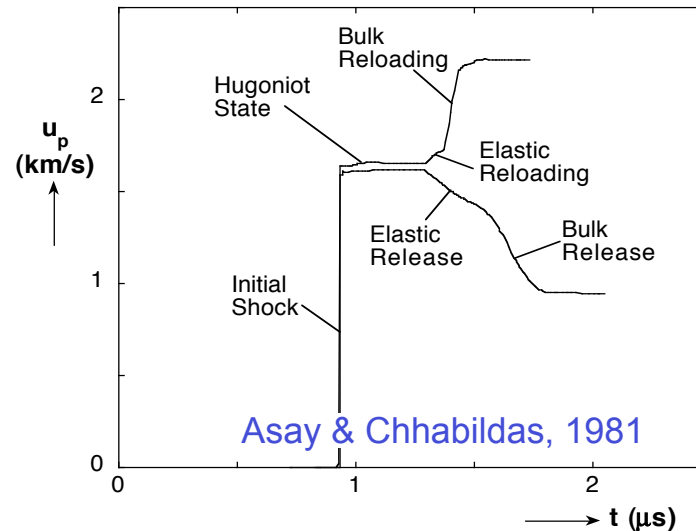


# Self Consistent Method (Asay, Lipkin, Chhabildas, *et al.*)

## reshock & release configuration



## results for 6061-T6 Al



- wave speed determined from VISAR release or reloading profiles
- unloading path calculated from incremental relations

*strength based on relative difference from Hugoniot state*



# Past Work: the Self Consistent Method

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Sandia - reshock/release of Al, Be, Cu, Mo, Ta, V, W  
(Asay, Chhabildas, *et al.*)

WSU - reshock/release of Al

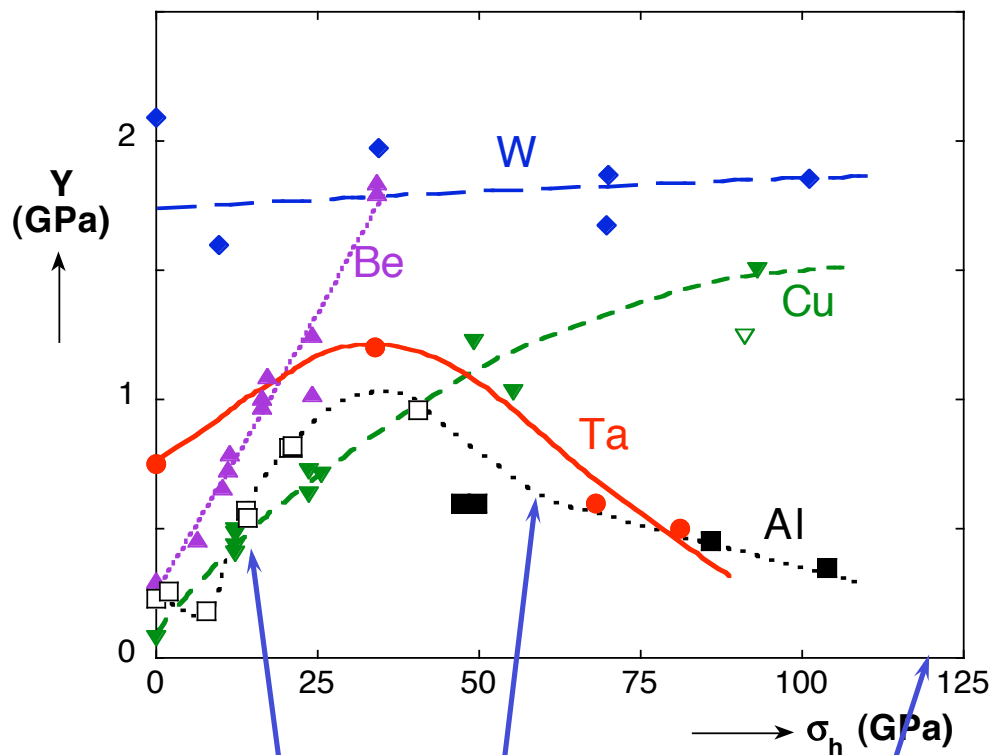
LANL - release of 2024 Al

Russia - results of an experiment by Al'tshuler *et al.*,  
1999, using longitudinal gauges agree with results  
of Asay & Chhabildas [1981] and with lateral gauge  
measurements



# Strength of Metals Under Shock Loading

metals studied to high pressure using self-consistent method



- strength increases due to work and pressure hardening
- ultimately decreases due to thermal softening

pressure and/or thermal work hardening

softening

melting



# Outline of Talk

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- Introduction to shock and high-pressure physics
- Gas gun technology and examples
- The Z machine and examples
- Background on high pressure strength and techniques to measure it
- **Strength of ceramics under shock loading**
- Strength of metals under isentropic loading
- Conclusions



# Strength of Armor Ceramics

- $\text{Al}_2\text{O}_3$ ,  $\text{B}_4\text{C}$ , and  $\text{SiC}$  considered or fielded for personnel or vehicle protection
- ceramic armor performance closely linked to specific configuration --> need for accurate models for use in armor design
- “conventional wisdom” says (most) ceramics lose strength when shocked



Reinhart, W. D. and L. C. Chhabildas (2003). "Strength properties of Coors AD995 alumina in the shocked state." *International Journal of Impact Engineering* **29**: 601-619.

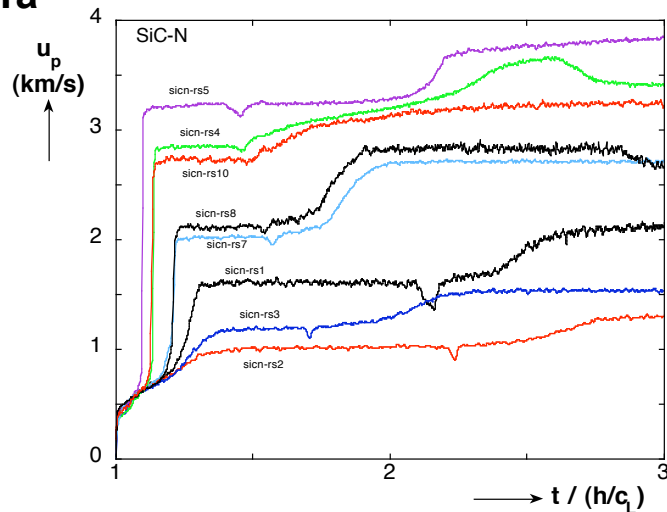
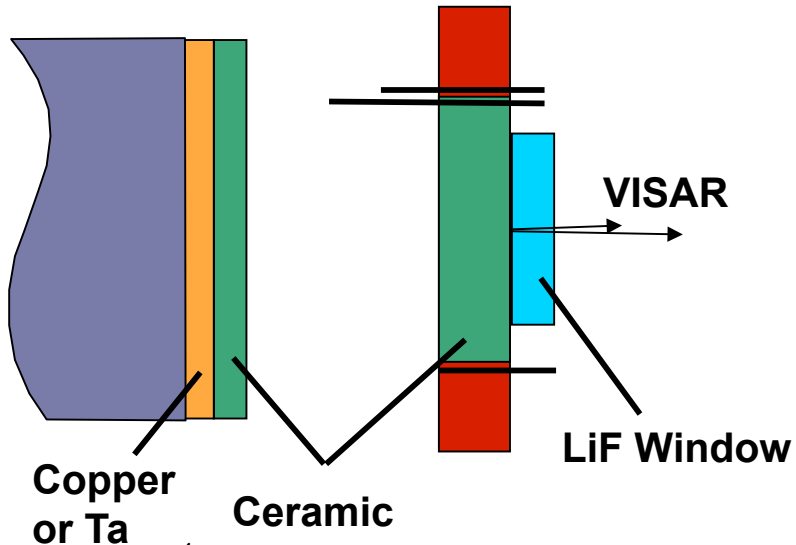
Vogler, T. J., W. D. Reinhart and L. C. Chhabildas (2004). "Dynamic behavior of boron carbide." *Journal of Applied Physics* **95**: 4173-4183.

Vogler, T. J., W. D. Reinhart, L. C. Chhabildas and D. P. Dandekar (2006). "Hugoniot and strength behavior of silicon carbide." *Journal of Applied Physics* **99**: 023512.

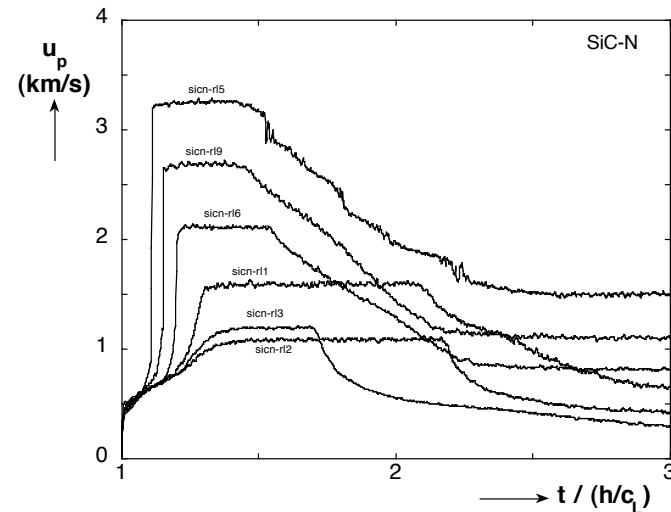
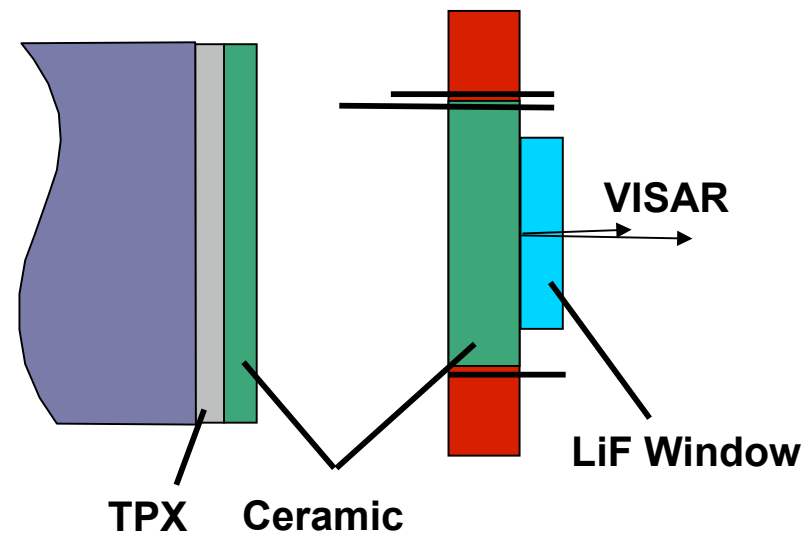


# Reshock & Release Experiments for $B_4C$ and SiC

## Reshock Configuration

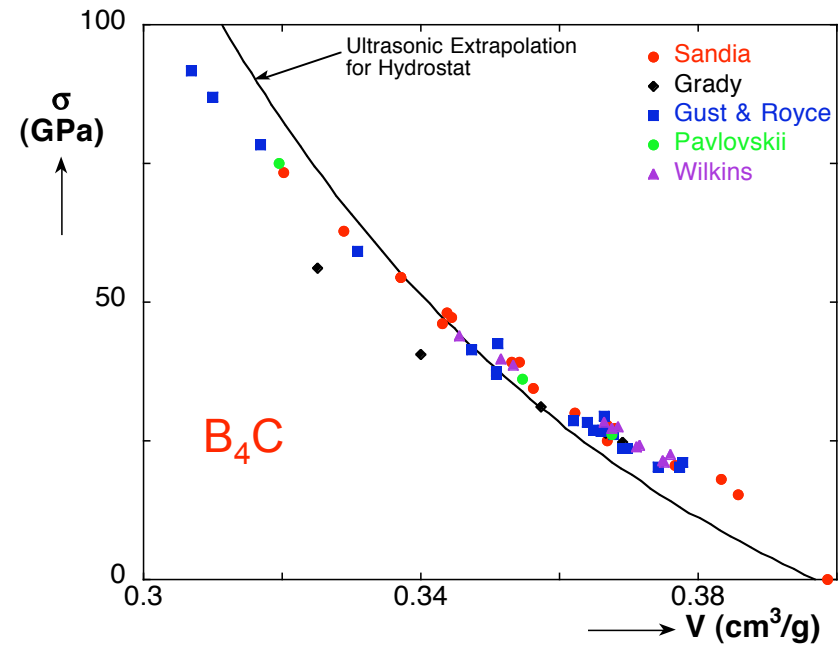
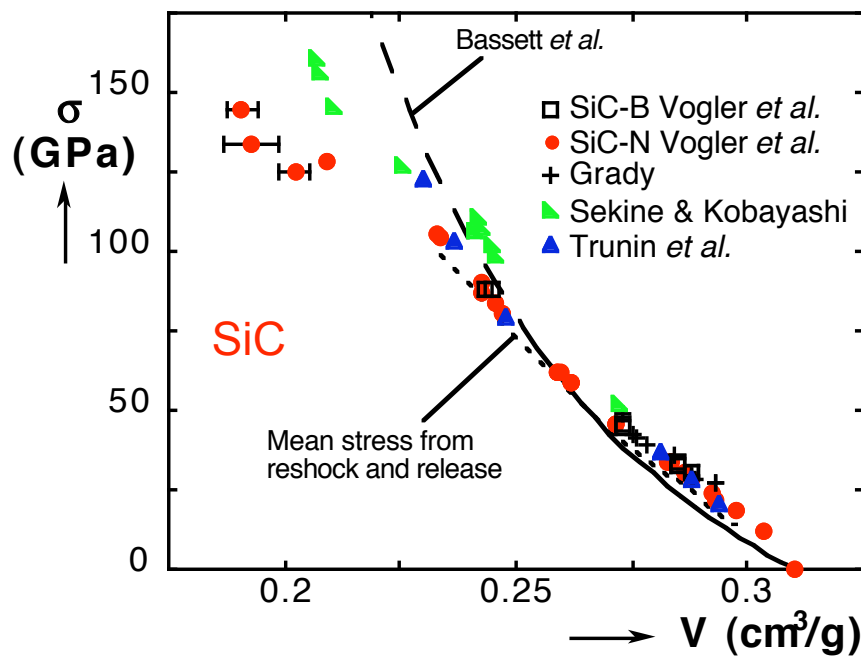


## Release Configuration





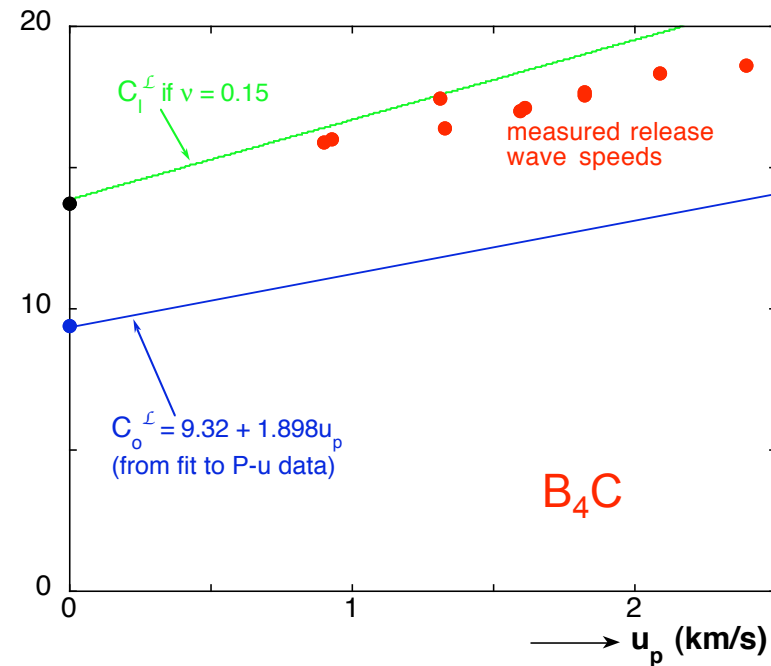
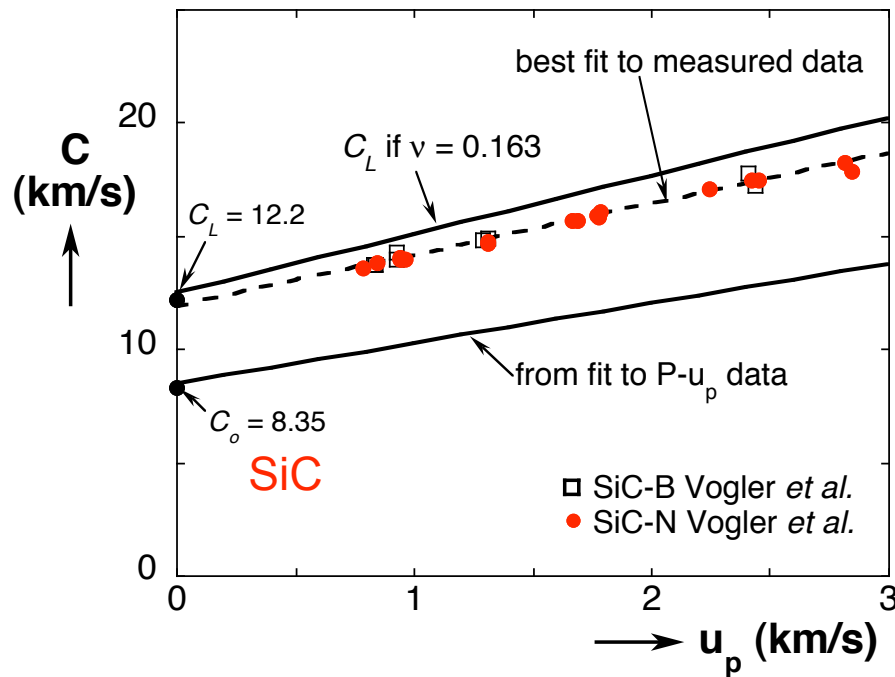
# Hugoniot for SiC and B<sub>4</sub>C



- Sandia studies have completed data sets, agreement with other studies generally good
- phase transition observed in SiC at ~105 GPa
- possible phase transition in B<sub>4</sub>C at ~40 GPa, but evidence inconclusive (behavior may be due to shock amorphization)



# High Release Wave Speeds Are Indicative of Elastic Behavior



- release waves arrive much faster than expected bulk wave speeds (based on Hugoniot)

- $v$  appears to be increasing in  $B_4C$

$$C_{bulk} \approx \frac{1}{\rho_o} \left. \frac{d\sigma}{du} \right|_{Hugoniot}$$

**material displays elastic behavior in shocked state**

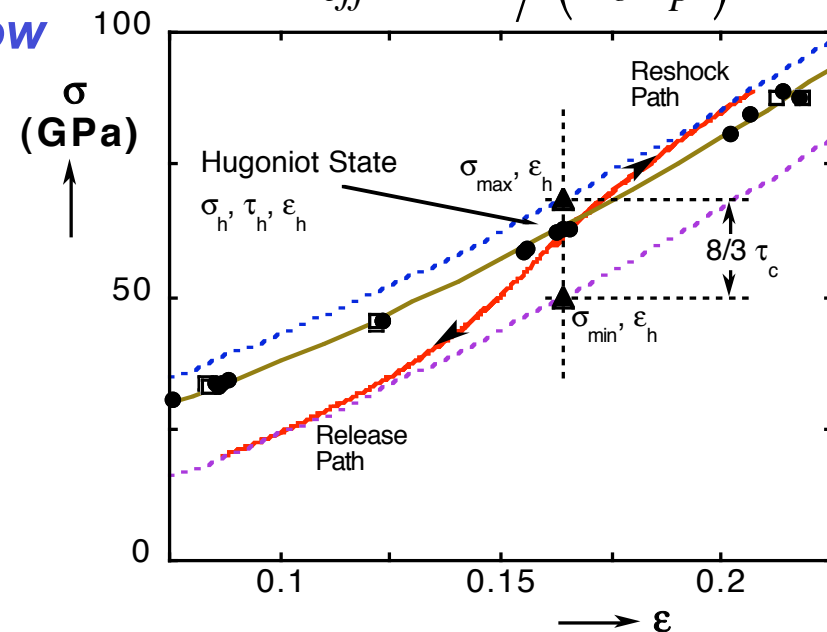
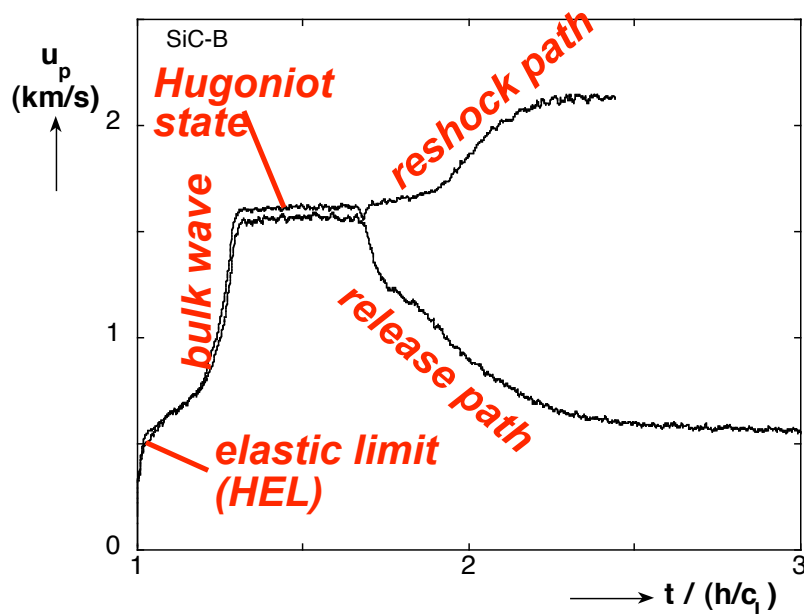


# Analysis of Reshock & Release Experiments

- *Incremental form of conservation equations used to calculate  $\sigma$ - $\epsilon$  paths*  
 $\Delta\sigma = \rho_o c \Delta u$   
 $\Delta\epsilon = \Delta u / c$
- *Account for presence of window*

*c based on effective shock:*

$$U_{eff} = \sigma^H / (\rho_o u_p^H)$$

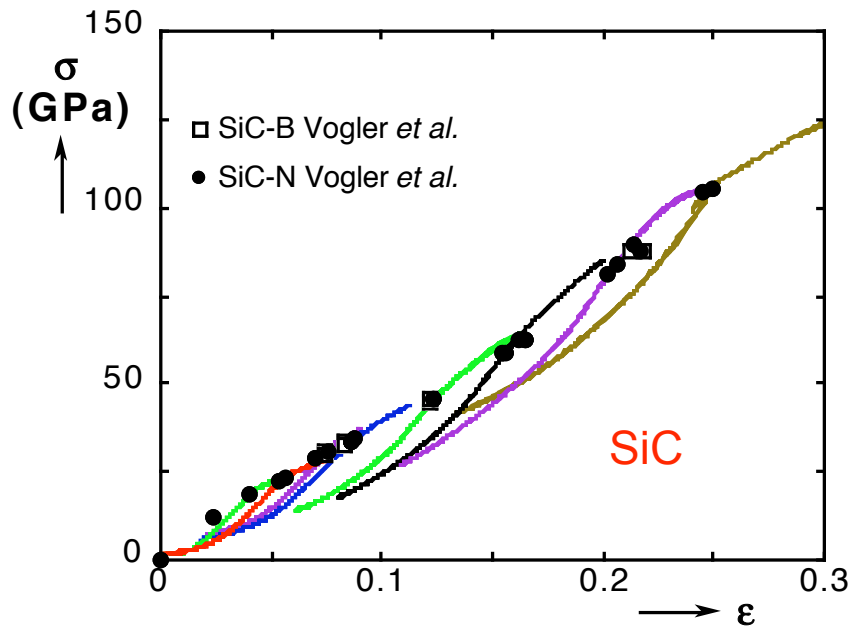


- *most of reshock path arrives later than release*
- *both arrive at elastic wave speed*

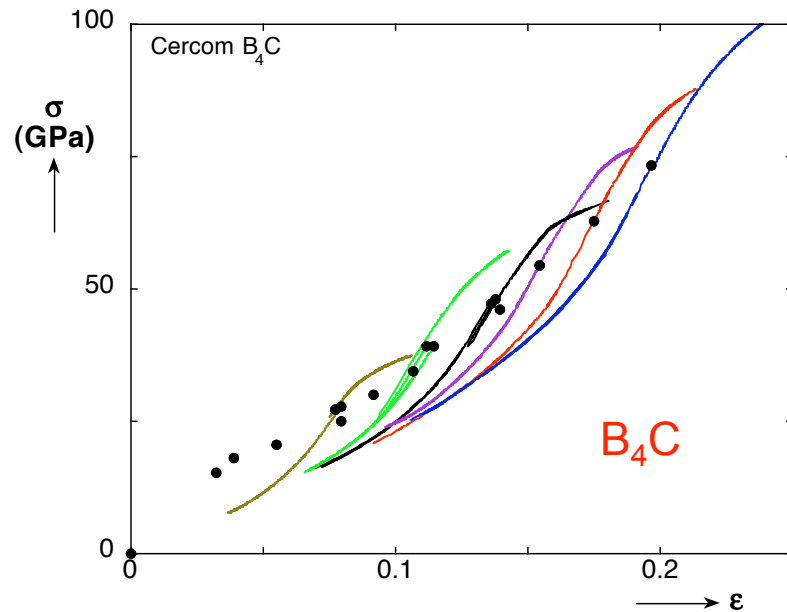
- *Technique to estimate:*
  - *Strength of the material in the shocked (damaged) state*
  - *Dynamic mean stress curve*
  - *Shear stress in Hugoniot state*



# Reshock & Release Paths



- release paths below Hugoniot
- reshock paths only slightly above Hugoniot
- shallow reshock  $>100$  GPa due to phase transformation

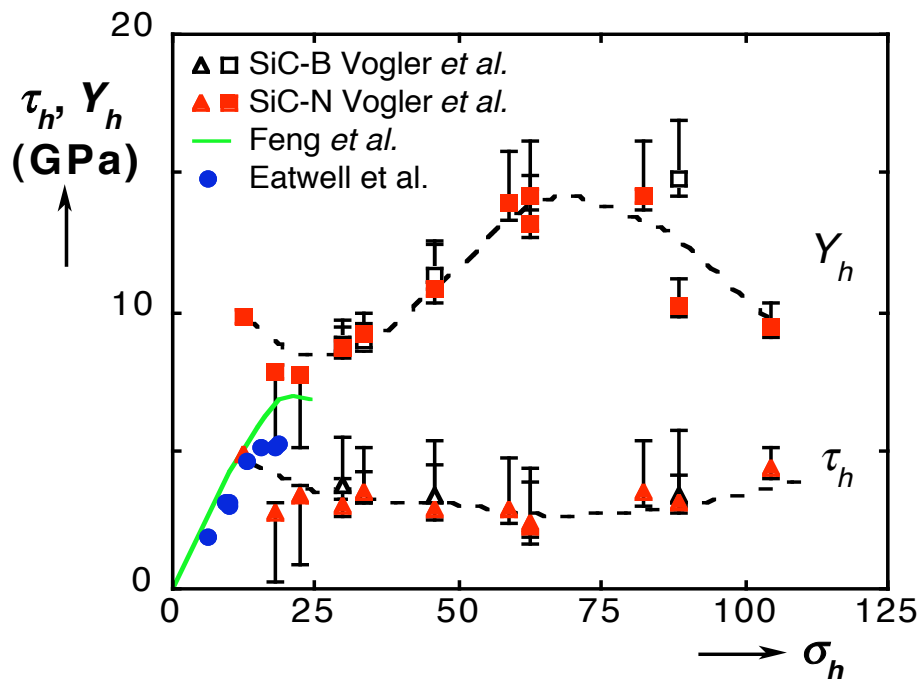


- release paths below Hugoniot
- reshock paths significantly above Hugoniot

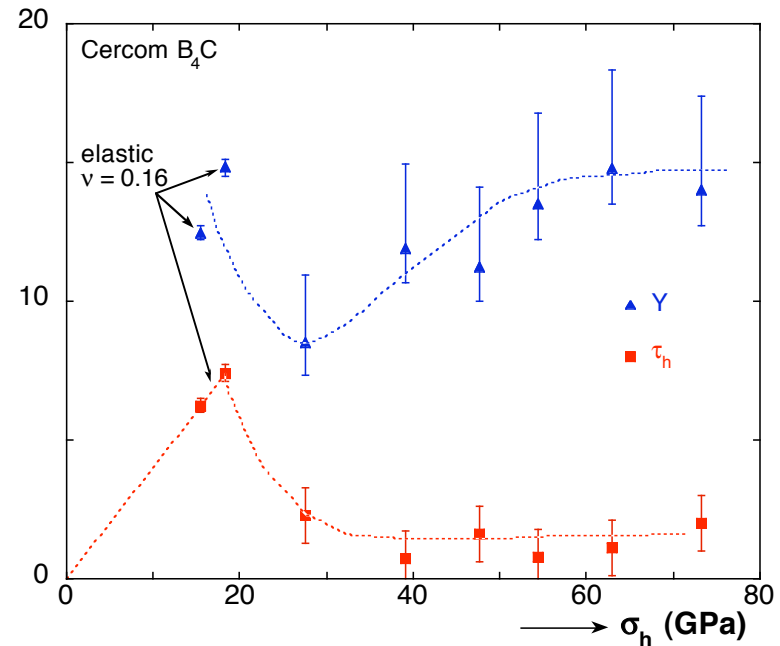
comparable behavior on release; SiC behaves in “metal-like” manner while  $B_4C$  behaves unexpectedly



# Strength in Shocked State



- significant strength retained
- apparent loss of strength approaching phase transition
- $\tau_h$  remains approximately constant



- material retains only small shear stress in Hugoniot state but large apparent strength during reshock/release
- strongly suggestive of damage
- kinetics and physical mechanisms unclear

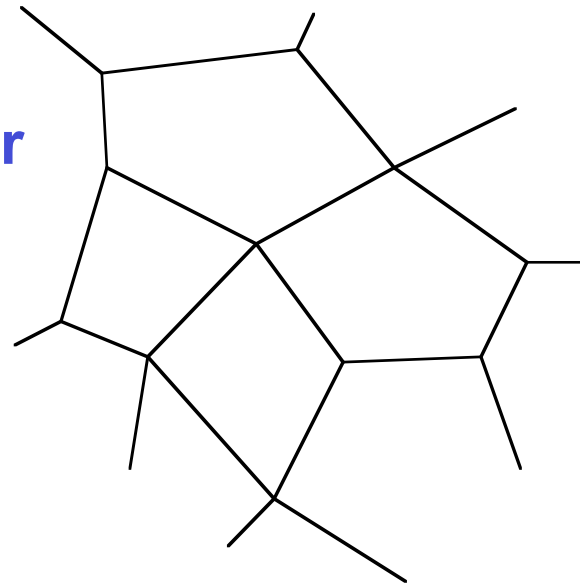
*how important is this behavior in an armor application?*



## Behavior of $B_4C$ (and $Al_2O_3$ ): A Possible Physical Explanation

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- initial shock creates damaged (micro-cracked) material and possible local melting but relatively small relative motion
- material equilibrates after shock passage; melted material solidifies; interfaces at rest or “healed” by solidification (similar to concept of Swegle & Grady, 1986)
- when perturbed by release or reshock, material responds elastically





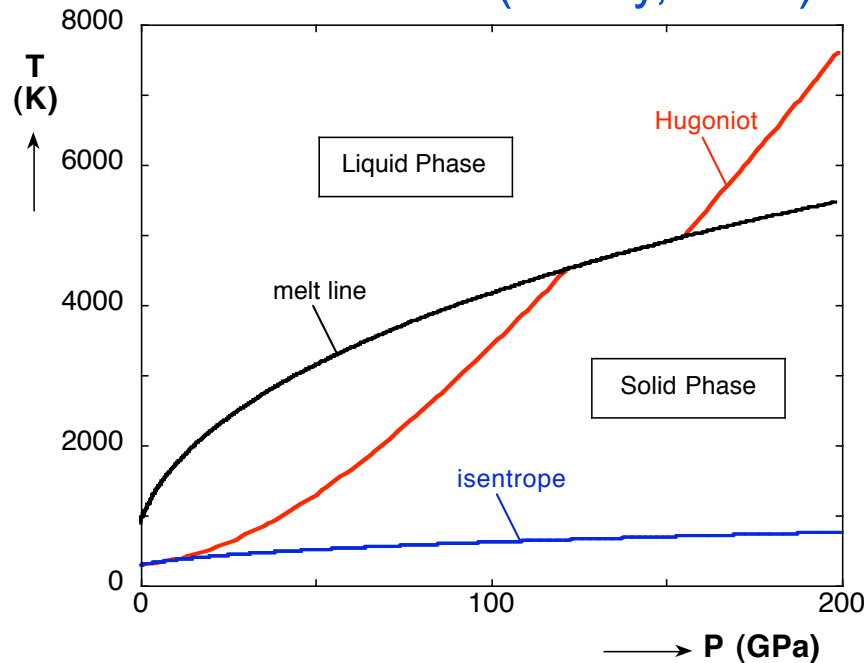
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- Strength of ceramics under shock loading
- **Strength of metals under isentropic loading**
- Conclusions

# Isentropic Loading Accesses Cool Regimes Where Strength Is More Important

Sesame 3700 (Kerley, 1987)



- Hugoniot passes into liquid phase at ~120 GPa
- Isentrope remains in solid phase, rise of T minimal

Steinberg-Guinan Strength Model (rate-independent version):

$$\frac{Y}{Y_o} = \left(1 + \beta(\varepsilon_p + \varepsilon_i)\right)^n \frac{G(P,T)}{G_o} \quad G(P,T) = G_o + \frac{\partial G}{\partial P} \frac{P}{\eta^{1/3}} + \frac{\partial G}{\partial T} (T - T_o)$$

**strength larger under isentropic loading due to smaller  $\Delta T$**



# Strength Under Quasi-Isentropic Loading: Previous Work

aluminum - isentrope stiffer than Hugoniot to 9 GPa (Barker-SNL)

tungsten - isentrope from graded-density impactor lies above Hugoniot up to 140 GPa due to strength

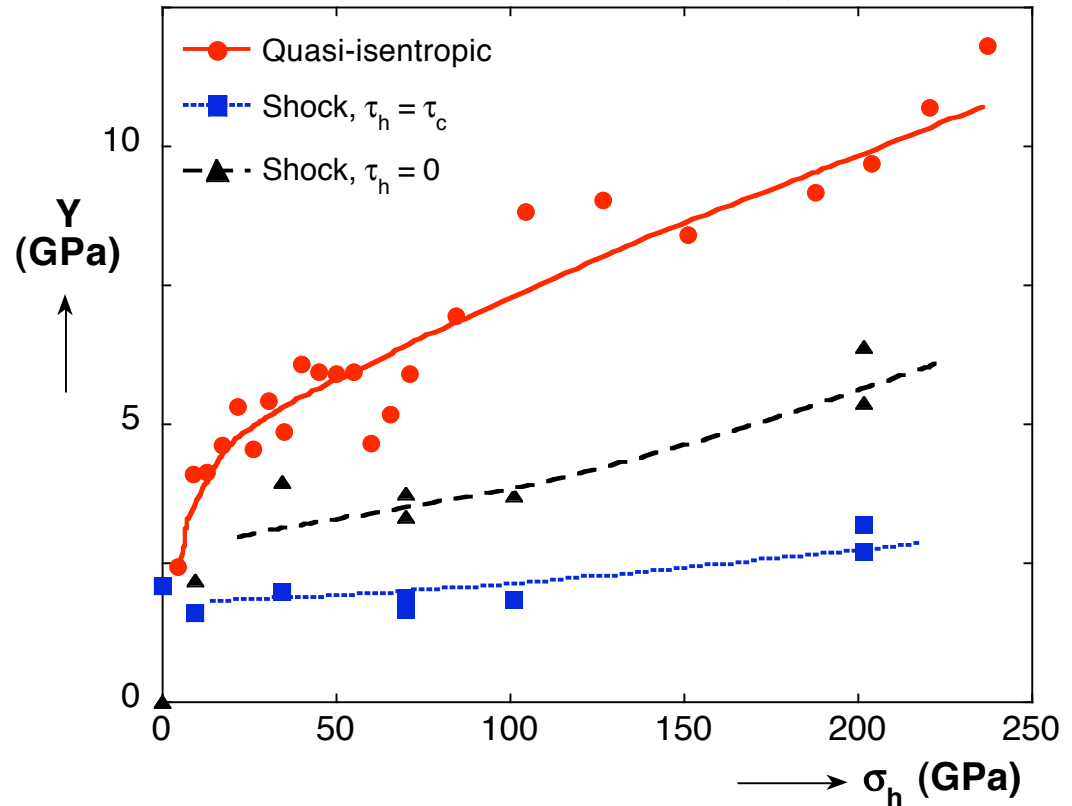
molybdenum - isentrope from Z compared with hydrostat to estimate strength (Reisman-LLNL/SNL)

lateral gauges

- AD-1 aluminum and copper (Bat'kov *et al.*)
- copper, iron, steel (Rosenberg *et al.*)

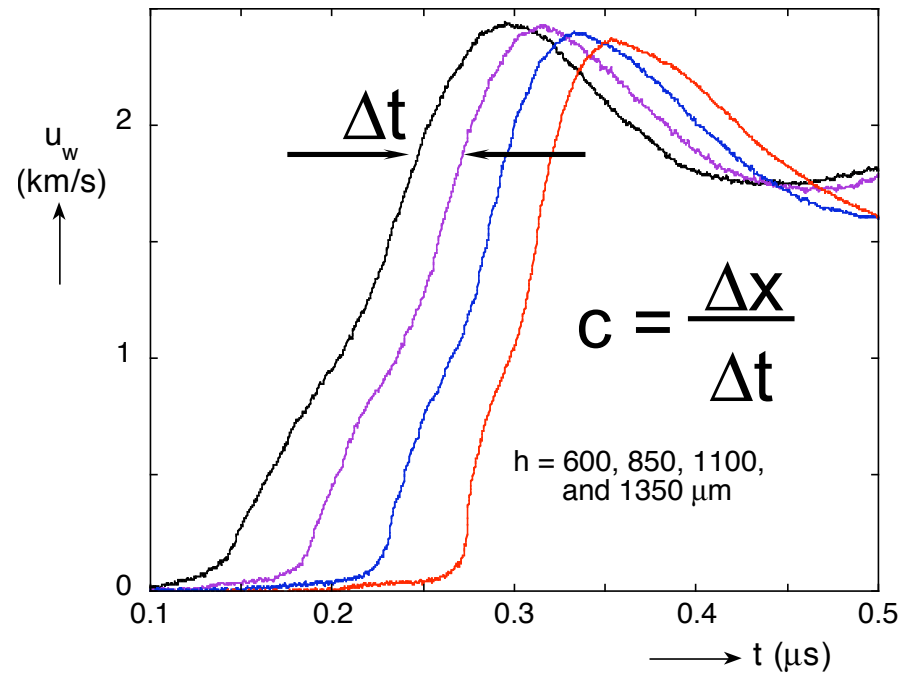
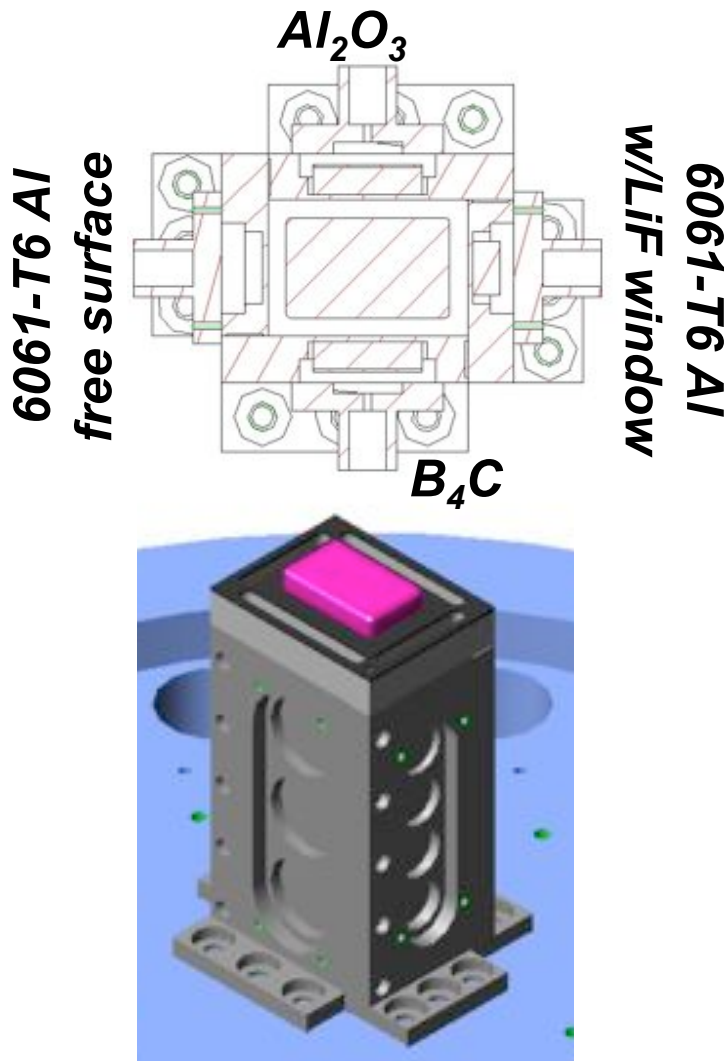
growth of Rayleigh-Taylor instabilities - aluminum and vanadium (LLNL)

Chhabildas & Barker, 1988





# Results for 6061-T6 Aluminum to 55 GPa



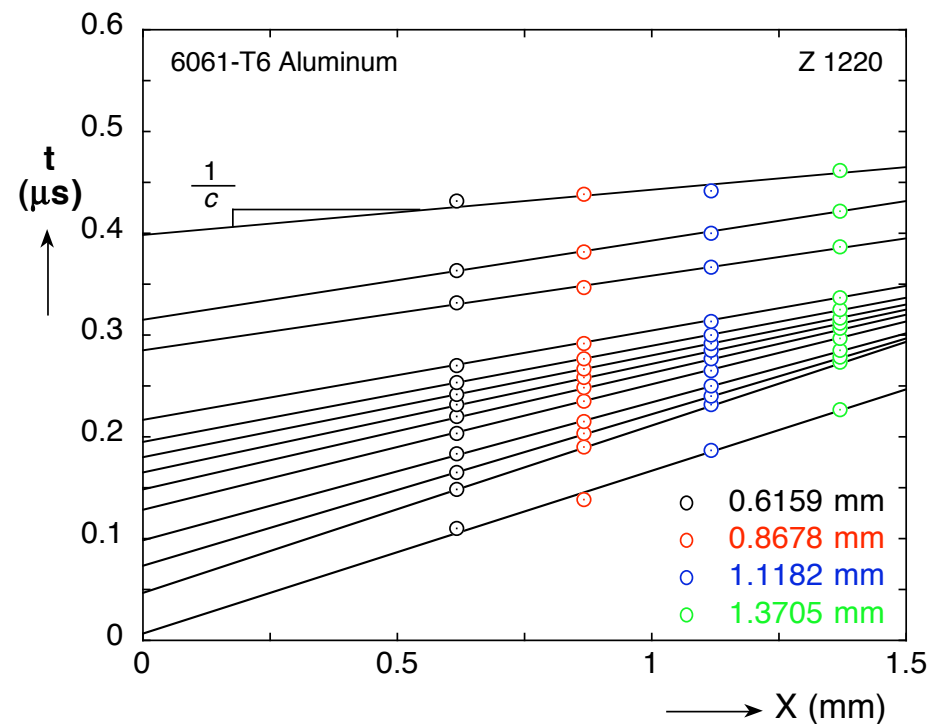
- four profiles, both loading and unloading histories
- peak attenuation evident
- unloading structure develops in thicker samples



# Lagrangian Analysis Technique

- *backwards integration technique of Hayes is non-unique for elastic-plastic materials*
- *Lagrangian analysis technique follows previous work by Grady and others*

1) determine  $c(u_w)$  by least-squares fit to VISAR data

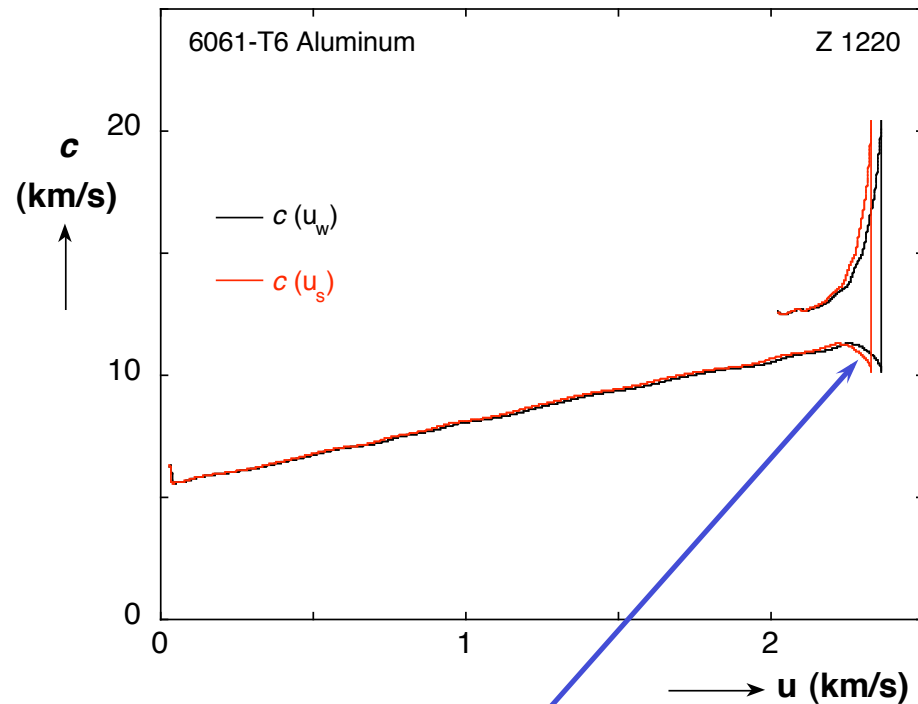




## Lagrangian Analysis Technique (2)

2) correct  $c(u_w)$  to  $c(u_p)$  by impedance matching of window and sample

$$\Delta u_s = \frac{Z_s + Z_w}{2Z_s} \Delta u_w$$



relaxation due to drop in strain rate?



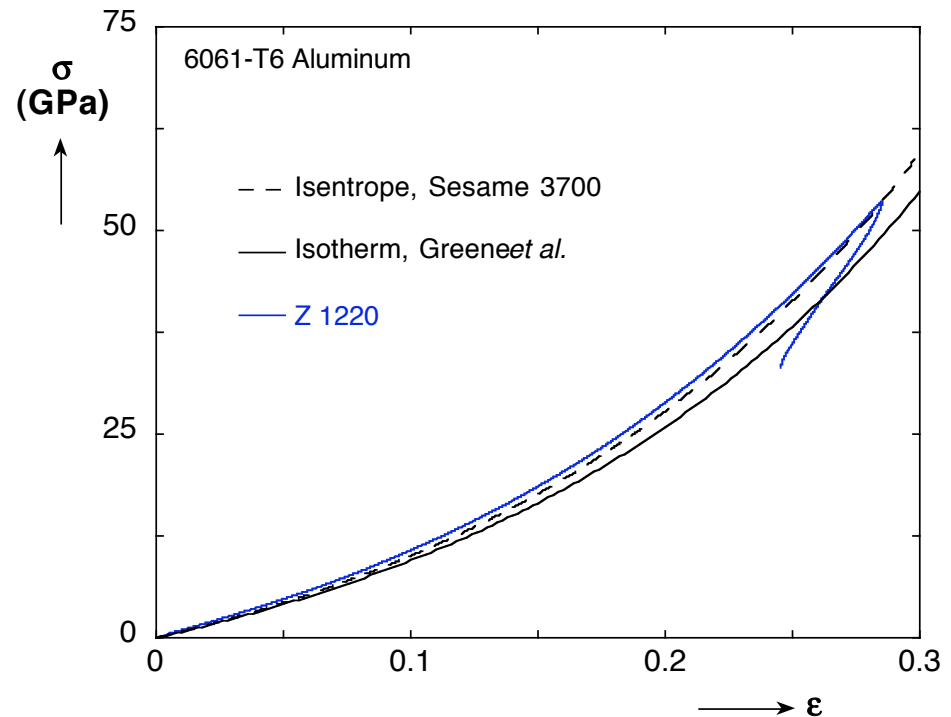
# Lagrangian Analysis Technique (3)

3) integrate stress and strain using incremental relations:

$$\Delta\sigma = \rho_o c \Delta u_p$$

$$\Delta\varepsilon = \Delta u_p / c$$

for unloading, ignore attenuation by beginning at lowest peak  $u_p$





# Assumptions in Lagrangian Analysis Technique

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- **characteristics not bent by window interactions**
- **rate-independent material**
- **all points experience same loading history**
- **window behavior known**
- **window loads along its principal isentrope**

**none of assumptions fully met!**

**none is too bad, either!**

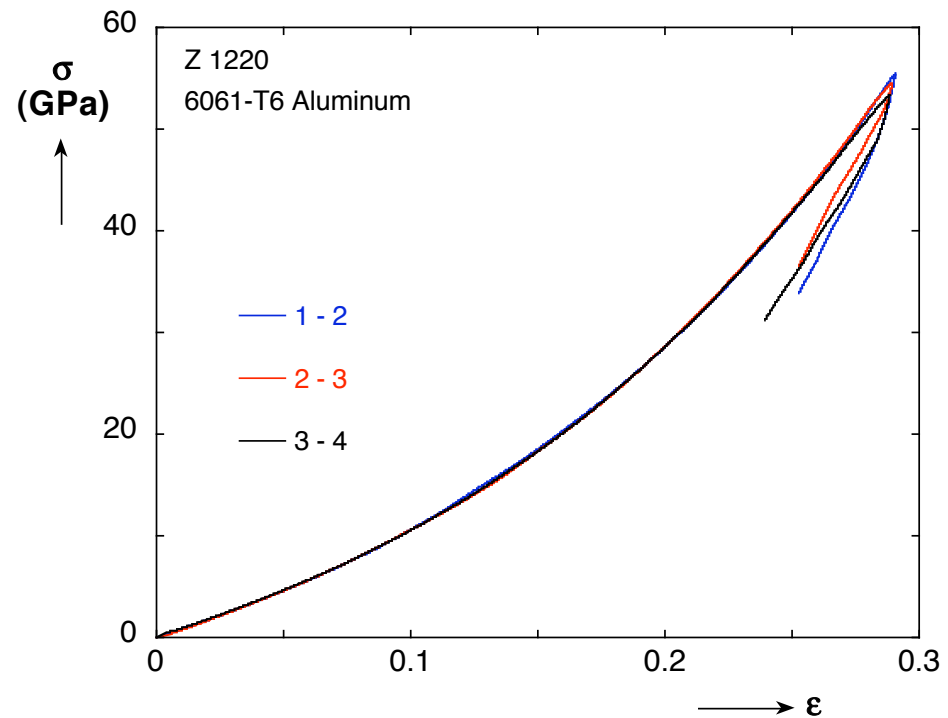


# Strength Measurement

Since stress-strain histories are somewhat different for each material point, VISAR results are analyzed in a pairwise fashion.

Loading responses are very similar for the three pairs.

Difference between loading and unloading curves is a measure of the strength.



$$Y = \frac{3}{4}(\sigma_L - \sigma_U)$$

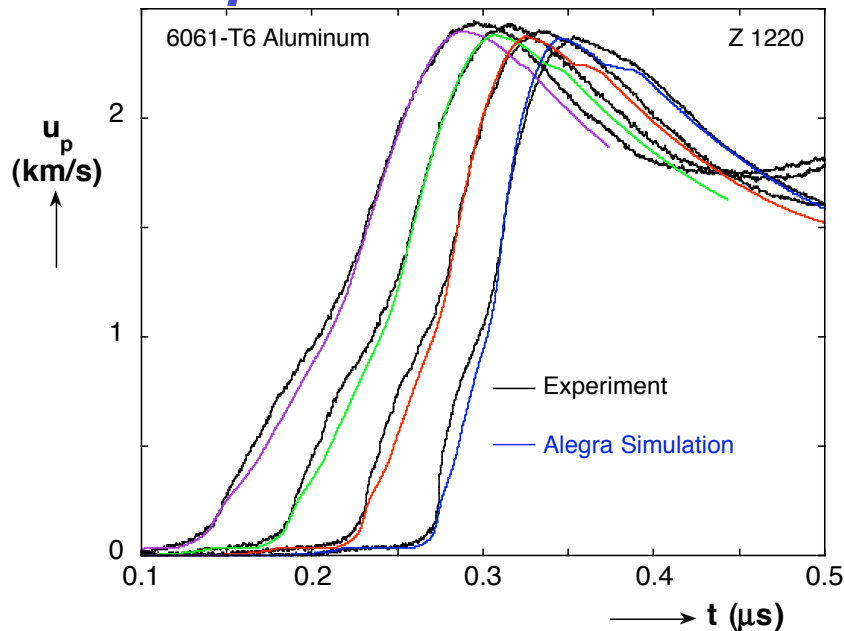
~~= 3.1, 2.2, 1.5 GPa~~

results suspect due to  
wave interactions



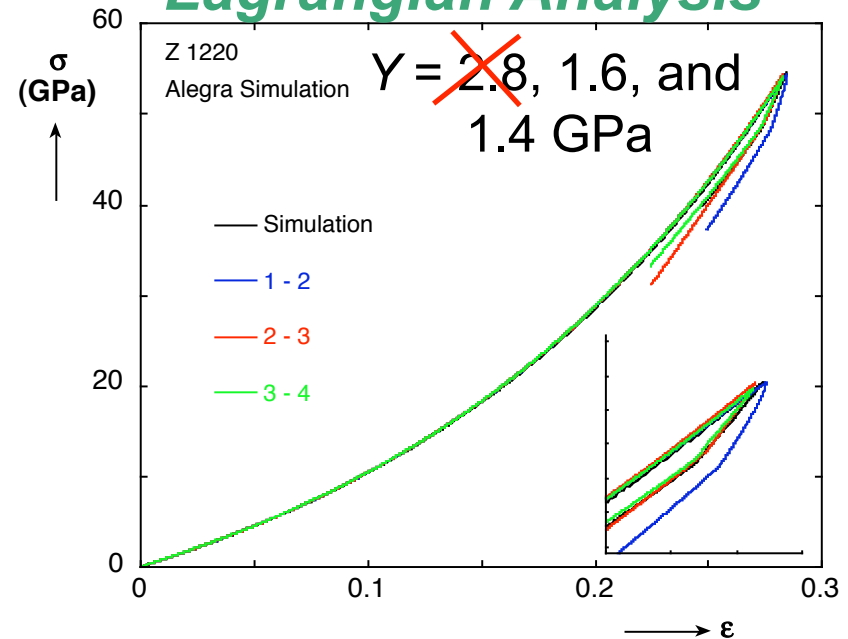
# Simulated Data for Z 1220

## Experiment & Simulation



- 1-D Alegra simulation - arbitrary Lagrangian-Eulerian magneto-hydrodynamics code, Steinberg-Guinan-Lund strength model
- current input somewhat off on loading, too much structure in unloading wave

## Lagrangian Analysis

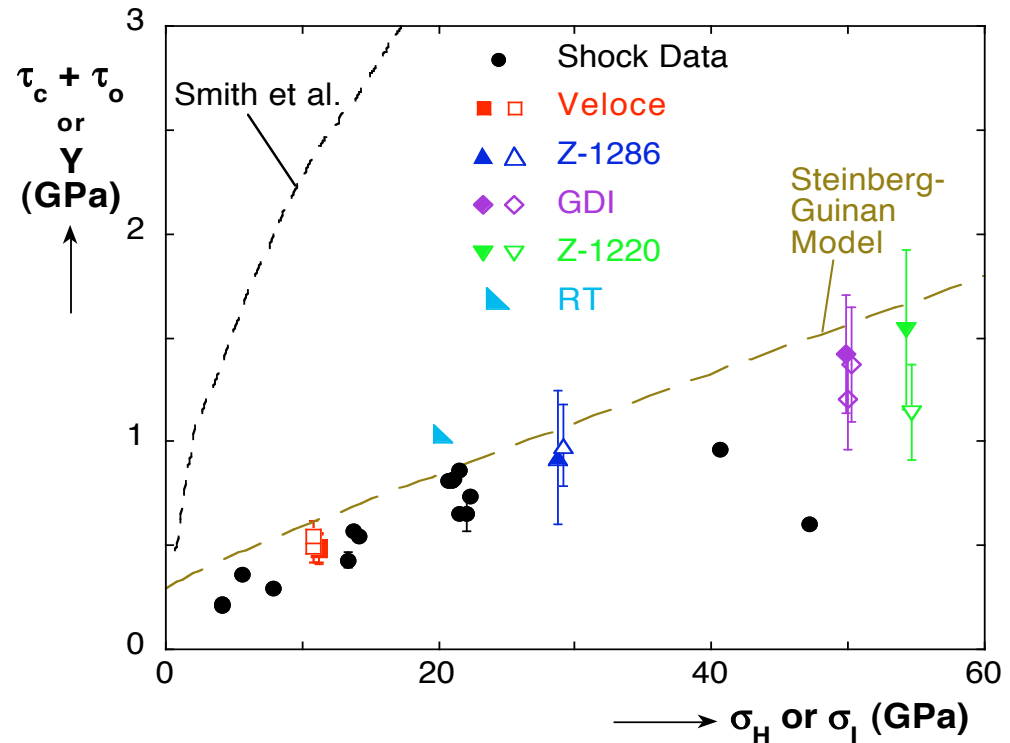


- loading captured extremely well
- first pair affected by window perturbations reaching drive surface, but others accurate measurements of  $Y$  (1.48 GPa)



# Strength Under Isentropic and Shock Loading

- strength under isentropic loading continues to increase with increased stress (due to pressure and/or work hardening)
- current experimental results agree with Steinberg model
- results of Smith et al. significantly different



Asay, J. R., Ao, T., Davis, J.-P., Hall, C. A., Vogler, T. J., and Gray, G. T. (2008). "Effect of initial properties on the flow strength of aluminum during quasi-isentropic compression," *J. Appl. Phys.* **103**, 083514.

Vogler, T. J., Ao, T., and Asay, J. R. (2009). "High-pressure strength of aluminum under quasi-isentropic loading," *Int. J. Plast.*, **25**, 671-694.



# Conclusions

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- self-consistent technique can be applied to shock or ramp loading for metals or ceramics
- SiC retains significant strength and shear stress in shocked state in a “metal-like” manner
- $B_4C$ 
  - retains considerable strength in the shocked state
  - shear stress is low or zero
  - current concepts of strength inadequate to describe this behavior
  - physical mechanisms unclear - damage or “thermal trapping”?
- strength measurements under isentropic loading to 60 GPa for aluminum
- strength results are relatively insensitive to timing errors (1 ns shift gives ~8% error in  $Y$  at 60 GPa)
- conservative estimate of 20-30% uncertainty on  $Y$  at 50 GPa for Al,



# Work Needed

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- comparison of different techniques (e.g. Rayleigh-Taylor, DAC, and self-consistent) for same materials
- utilize pressure-shear techniques to decouple effects of pressure, strain rate, work hardening, etc.
- grain scale investigations to understand mechanisms associated with strength retention/loss in metals and ceramics
- establish connections between experiments and molecular dynamics simulations
- investigate effect of solid-solid phase transformations on strength and vice-versa