

Strength of Materials at High Pressures and Strain Rates

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- Staff - C.S. Alexander, T. Ao, J.-P. Davis, D. Dolan, M. Furnish, M. Knudson, W.D. Reinhart, S. Root, J. Wise
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- STAR - T. Thornhill, J. Martinez, R. Palomino, H. Anderson
- Z - the large crew for design, assembly and operations
- retired colleagues: Lalit Chhabildas, Dennis Grady, Jim Asay

*numerous technicians and support staff
are essential to facility operations*



Outline of Talk

- **Introduction to shock and high-pressure physics**
- **Gas gun technology and examples**
- **The Z machine and examples**
- **Background on high pressure strength and techniques to measure it**
- **Strength of ceramics under shock loading**
- **Strength of metals under isentropic loading**
- **Conclusions**

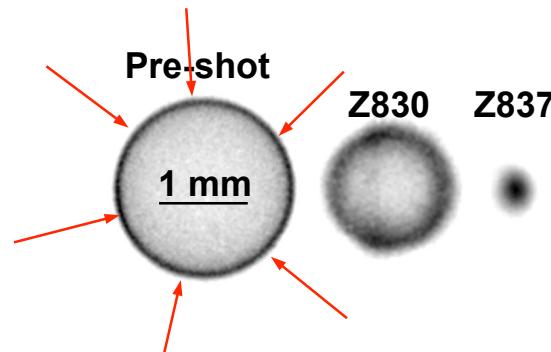


Why Do We Need To Know the Behavior of Materials Under Extreme Conditions?

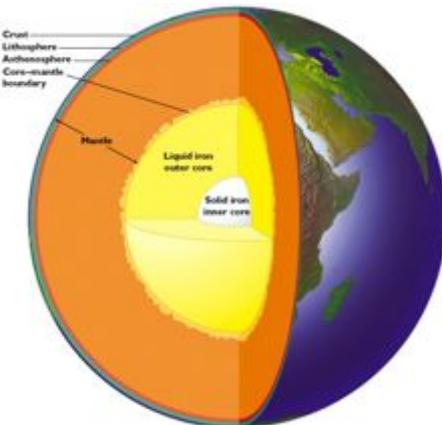
- weapons applications (warheads, armor, etc.)
- explosives behavior and applications



- inertial confinement fusion



- missile intercept; impact of asteroids or orbital debris ($V=8-15$ km/s)
- planetary science ($P \sim 360$ GPa, $T \sim 7000$ K)

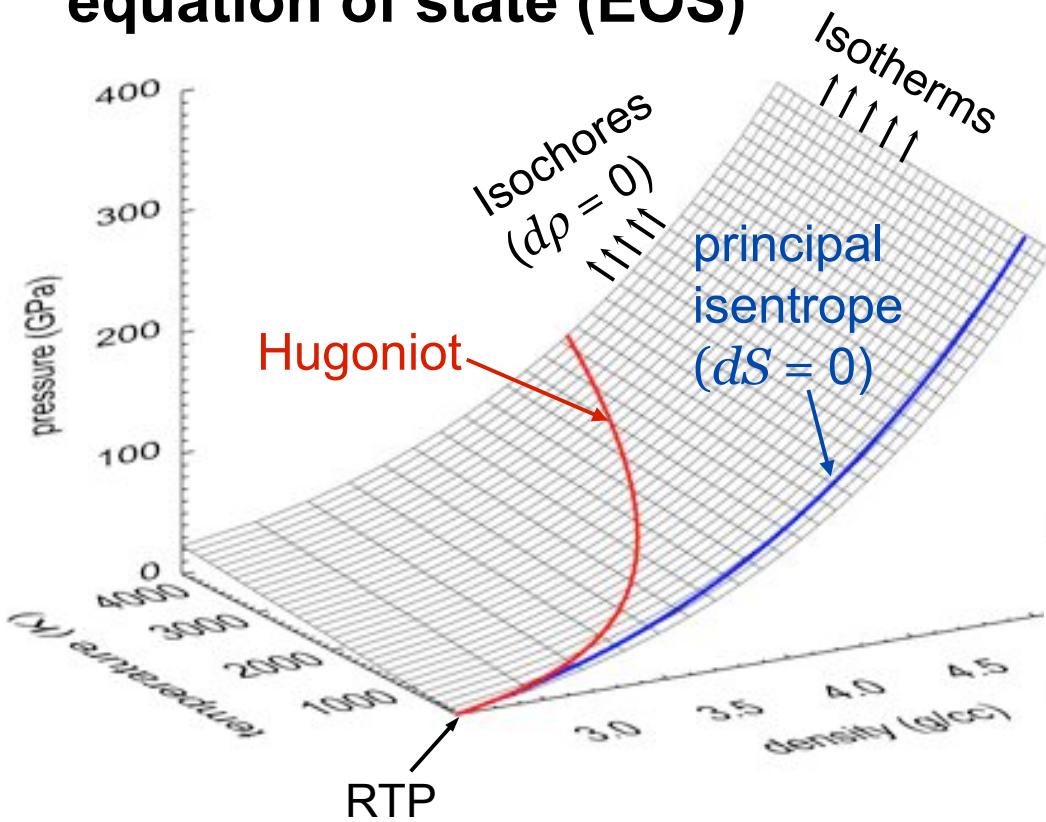




Material Behavior: EOS & Constitutive Aspects

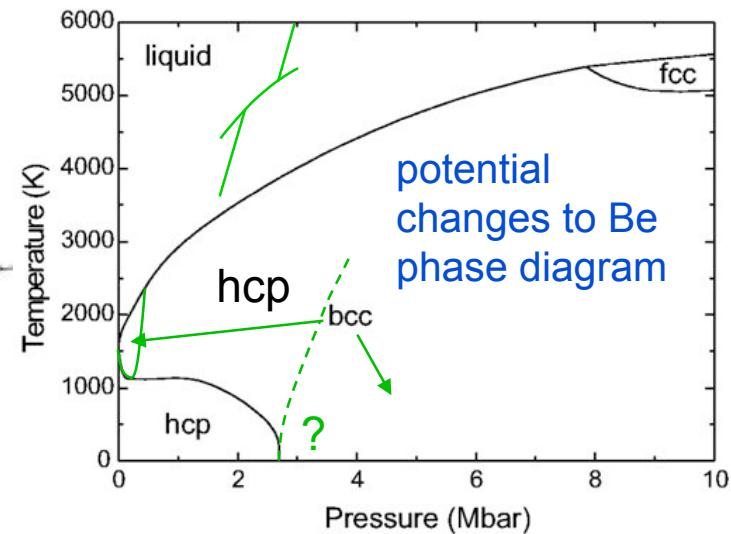
M. Knudson,
M. Desjarlais

equation of state (EOS)



one thermodynamic state variable as a function of two others:

pressure $P = P(r, T)$
Helmholtz energy $f = f(v, T)$

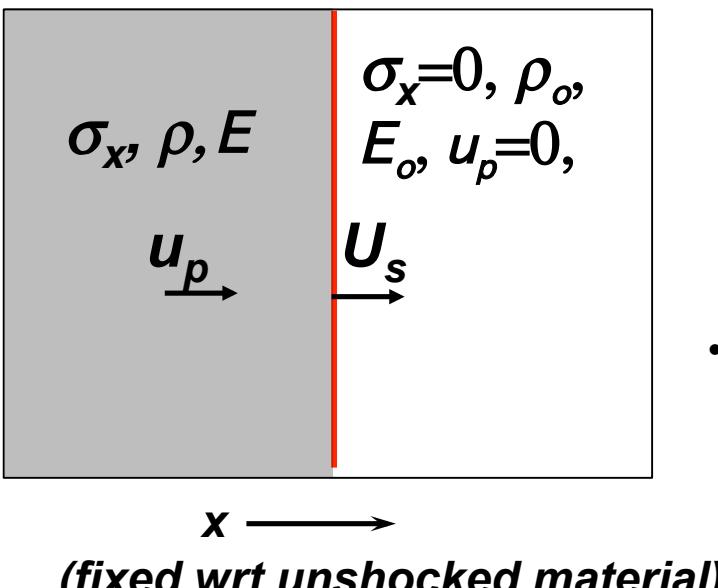


Also: strength, damage, spall (tensile failure), compaction



What is a Shock Wave?

- A “discontinuous” wave that moves at a fixed velocity (if steady)
 - wave front moves at speed U_s (*shock velocity*)
 - shocked material moves at speed u_p (*particle or mass velocity*)
 - uniaxial strain condition ($\varepsilon_y = \varepsilon_z = \varepsilon_{xy} = \varepsilon_{yz} = \varepsilon_{xz} = 0$)
- **shocked material** **unshocked material**
- States ahead and behind shock assumed to be in thermodynamic equilibrium
 - well defined temperature in each state
 - described by equilibrium thermodynamics
- Shock compression is adiabatic
 - very fast process (< 1 ns)
 - irreversible (i.e. NOT isentropic)
 - temperature *typically* increases



σ_x, ρ, E
 $u_p \rightarrow$

$\sigma_x=0, \rho_o,$
 $E_o, u_p=0,$
 $U_s \rightarrow$

$x \longrightarrow$
(fixed wrt unshocked material)



Conservation Equations and the Shock Hugoniot

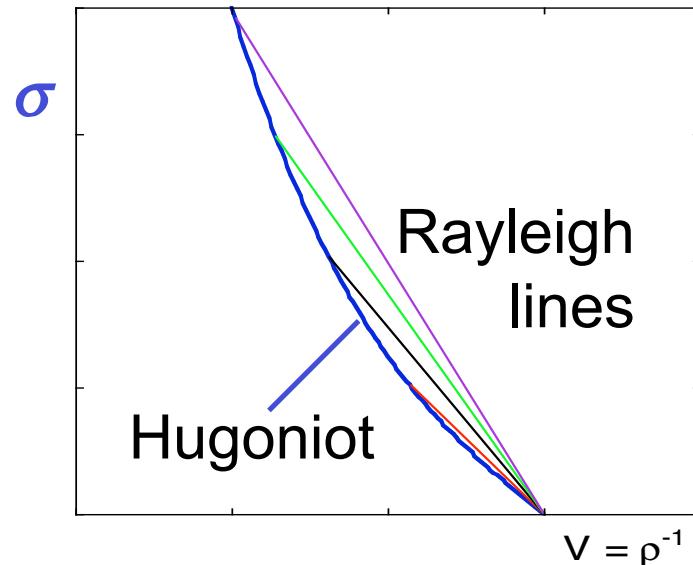
- Five variables: σ_x , u_p , U_s , ρ , and E
- Three conservation relationships (Rankine-Hugoniot jump conditions)
 - By measuring two variables (typically σ_x , u_p , or U_s), the other three can be determined

conservation of

mass: $\rho_0 U_s = \rho (U_s - u_p)$

momentum: $\sigma_x = \rho_0 U_s u_p$

energy: $E - E_0 = 0.5 \sigma_x (V_0 - V)$



material loads along the Rayleigh line, so the Hugoniot is a collection of end states, not a material response curve
the Hugoniot is not a complete equation of state (EOS)!



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Gas Guns to Generate Shock Waves



Single Stage Gun 100mm



~1 km/s
~30 GPa

Propellant Gun 89mm



~2 km/s
~100 GPa

Two-Stage Gun 29mm



~8 km/s
~700 GPa

gas guns

- launch thin plates (mm's) at high velocities
- well-posed, repeatable initial conditions
- sample is in uniaxial **strain**
- used to study material behavior at high pressures and strain rates
- usable in laboratory setting

Three-Stage Gun 17mm



~16 km/s
~2 TPa

Chhabildas, L. C., Dunn, J. E., Reinhart, W. D., and Miller, J. M. (1993). "An impact technique to accelerate flier plates to velocities over 12 km/s," *Int. J. Impact Eng.* **14**, 121-132.

also: explosives, lasers, magnetic loading (Z)

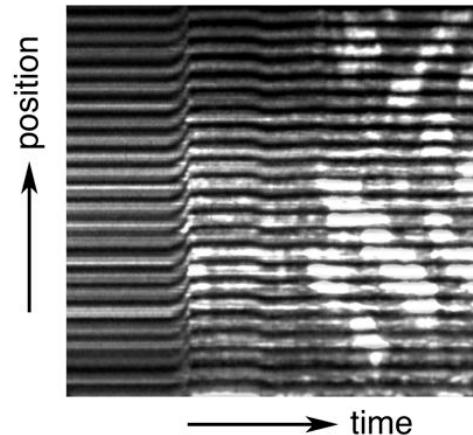


Diagnostics for Dynamic Experiments

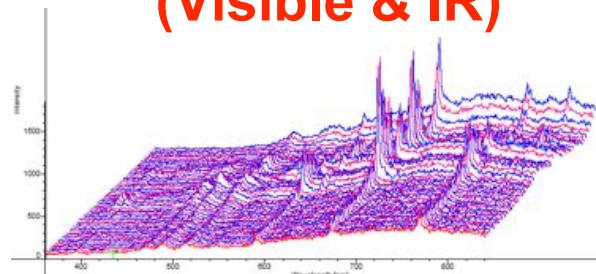
Velocity Interferometry (VISAR & PDV)



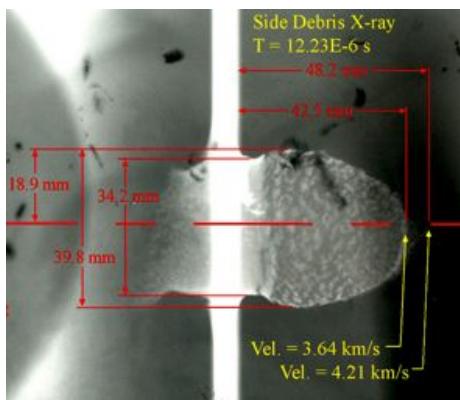
Line-VISAR



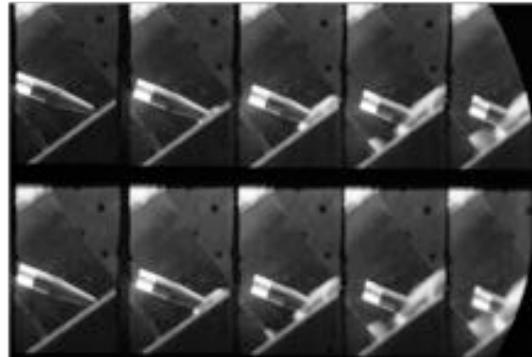
Time-Resolved Spectroscopy (Visible & IR)



Flash X-rays



High-Speed Photography



Pressure Gauges

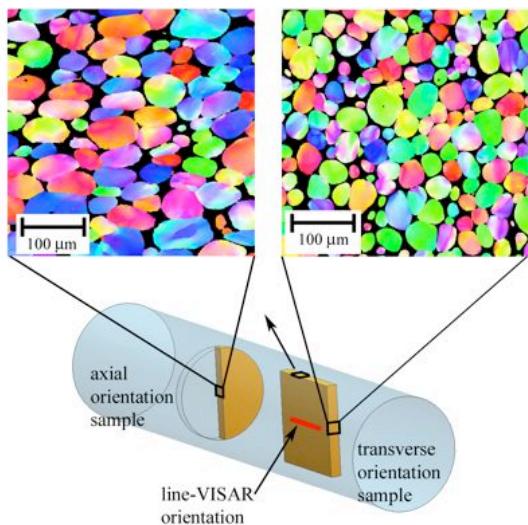


Advanced Diagnostics: pRad, synchrotron, etc.

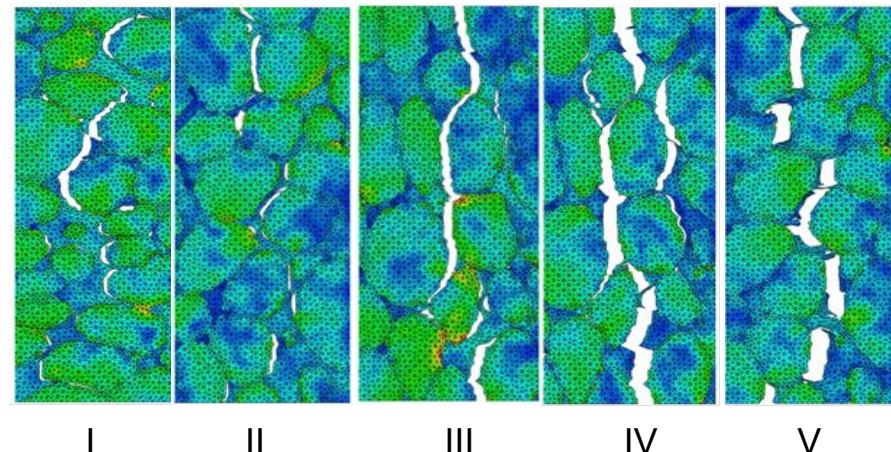
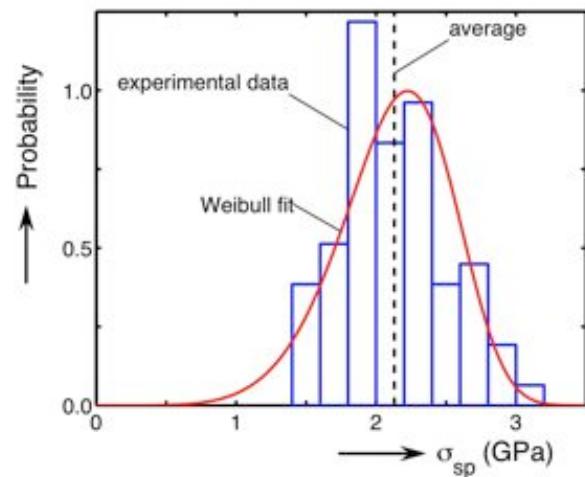
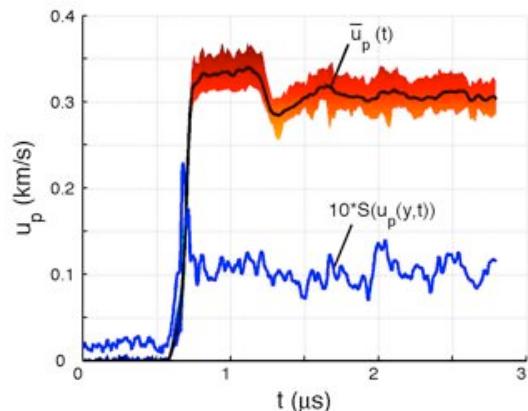
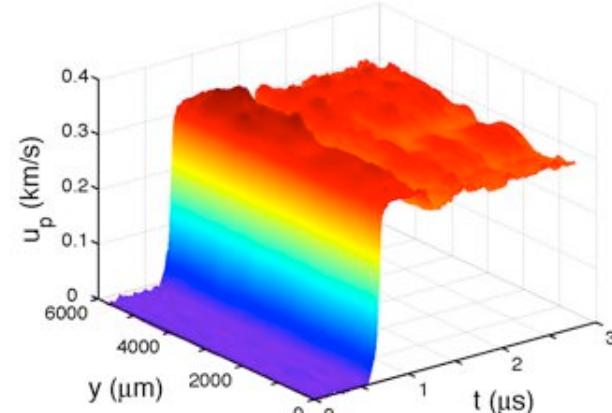


Heterogeneous and Statistical Aspects of Spall

J. Clayton,
T. Vogler



WHA - tungsten heavy alloy

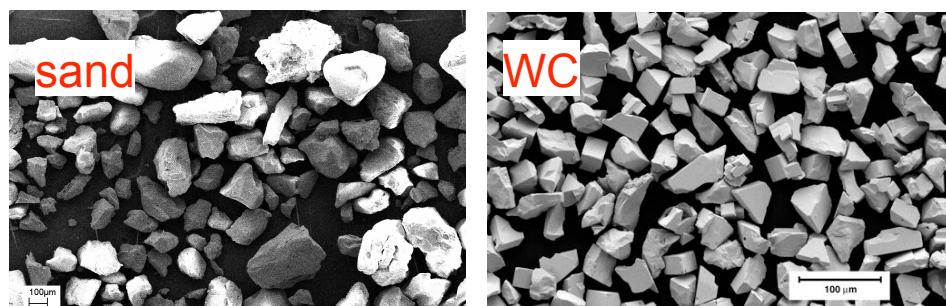
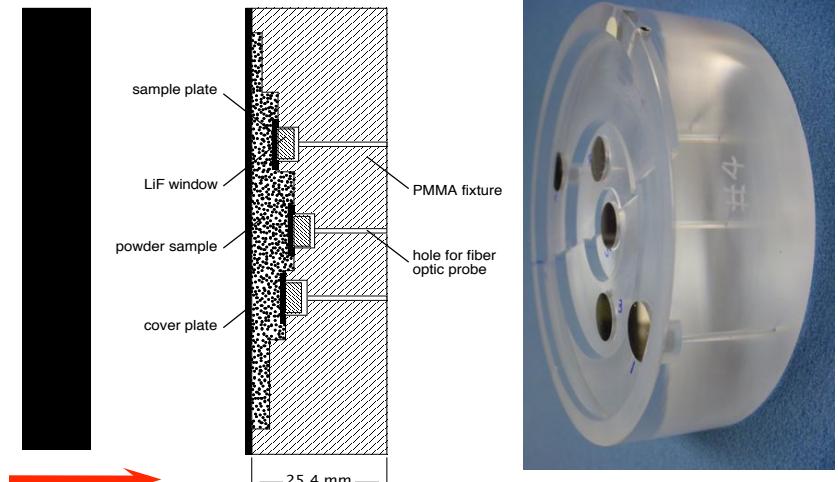


Vogler, T. J., and Clayton, J. D. (2008). "Heterogeneous deformation and spall of an extruded tungsten alloy: plate impact experiments and crystal plasticity modeling," *J. Mech. Phys. Solids* 56, 297-335.



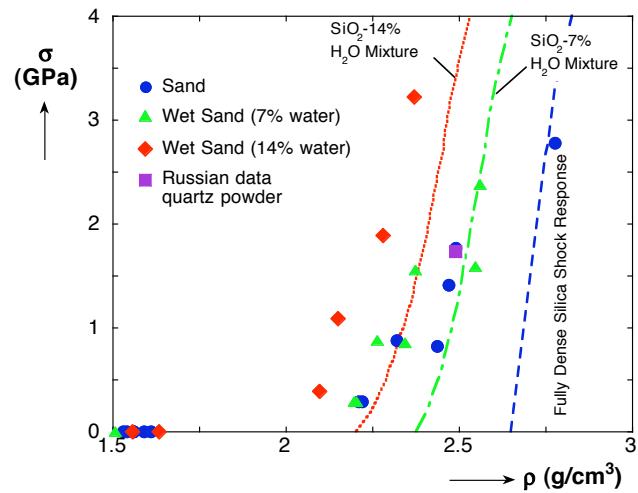
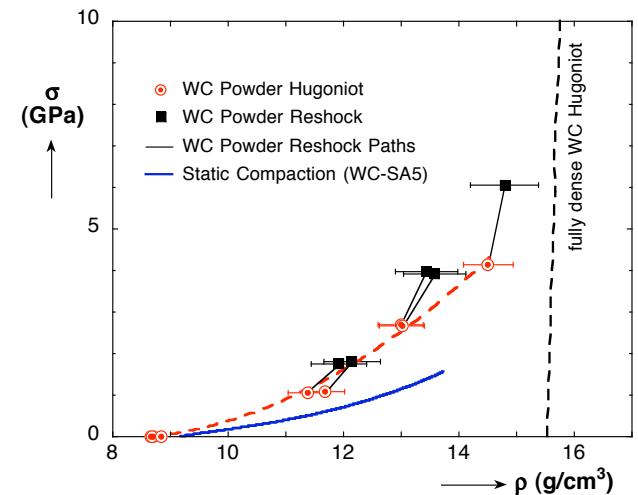
Compaction of Granular Ceramics

D. Grady,
T. Vogler



Vogler, T.J., Lee, M.Y., Grady, D.E., 2007. "Static and dynamic compaction of ceramic powders." *International Journal of Solids and Structures* **44**, 636-658.

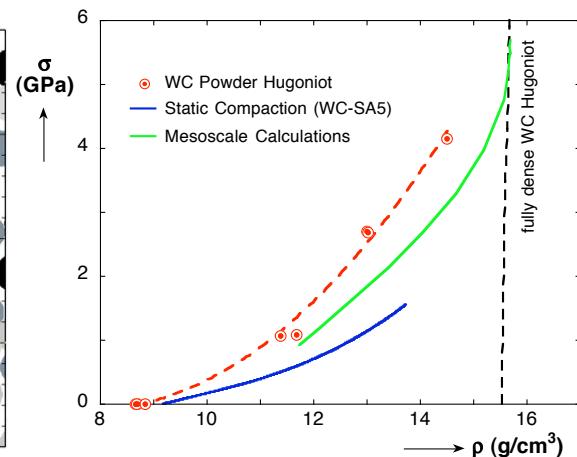
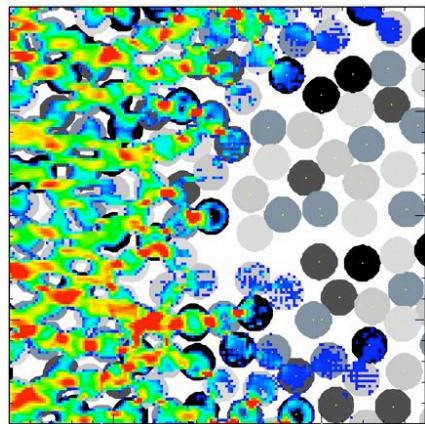
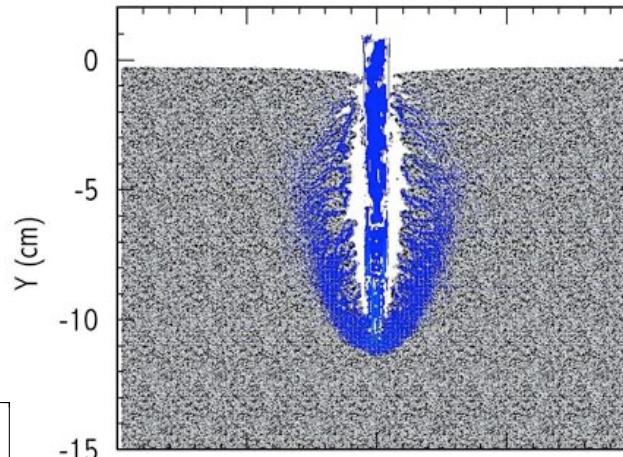
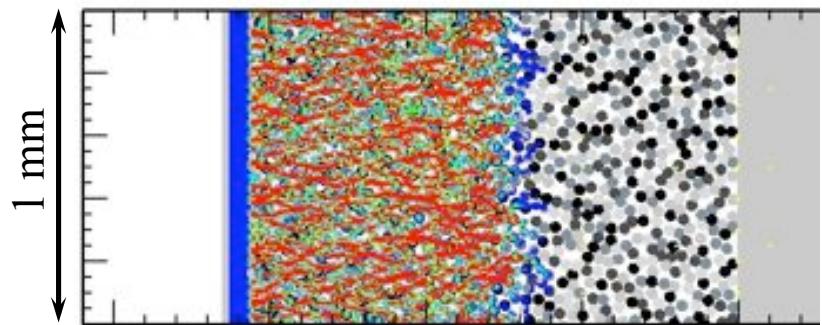
Brown, J.L., Thornhill, T.F., Reinhart, W.D., Chhabildas, L.C., Vogler, T.J., 2007. "Shock response of dry sand." in Shock Compression of Condensed Matter – 2007, American Institute of Physics, 1363-1366.



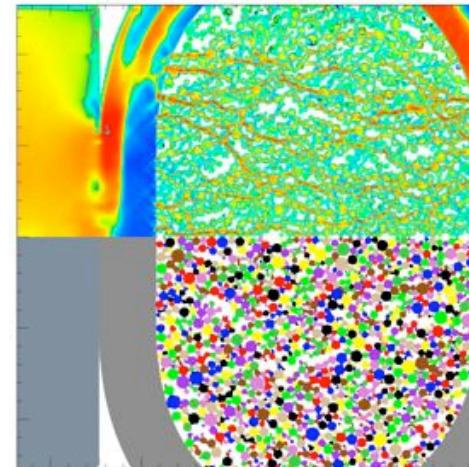


Mesoscale Modeling of Granular Materials

J. Borg,
T. Vogler



Borg, J. P., and Vogler, T. J. (2008). "Mesoscale simulations of a dart penetrating sand," *Int. J. Impact Eng.* **35**, 1435-1440.



Borg, J. P., and Vogler, T. J. (2008). "Mesoscale calculations of the dynamic behavior of a granular ceramic," *Int. J. Solids Structures* **45**, 1676-1696.

Borg, J. P., and Vogler, T. J. (2009). "Aspects of simulating the dynamic compaction of a granular ceramic," *Modelling Simul. Mater. Sci. Eng.* **17**, 045003.

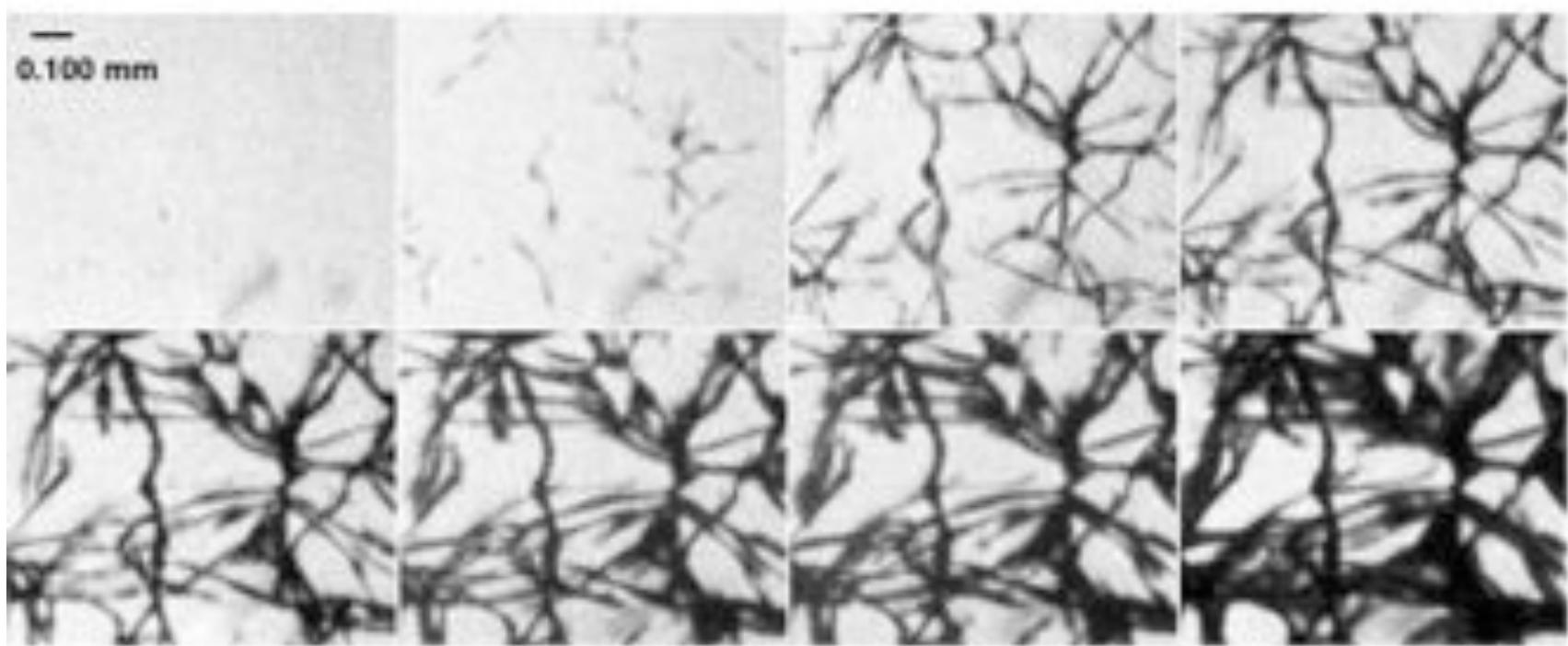


Nanosecond Freezing of Water

- 2.7 GPa (ice VII)
- 25 ns exposure

**$t(\text{ns})$ from
shock arrival =**

230	330	530	630
730	830	930	1530



D.H. Dolan and Y.M. Gupta, J. Chem. Phys. **121**, 9050 (2004).



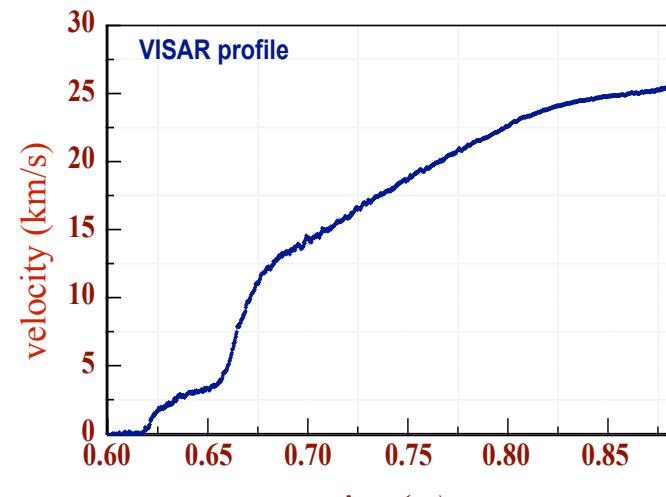
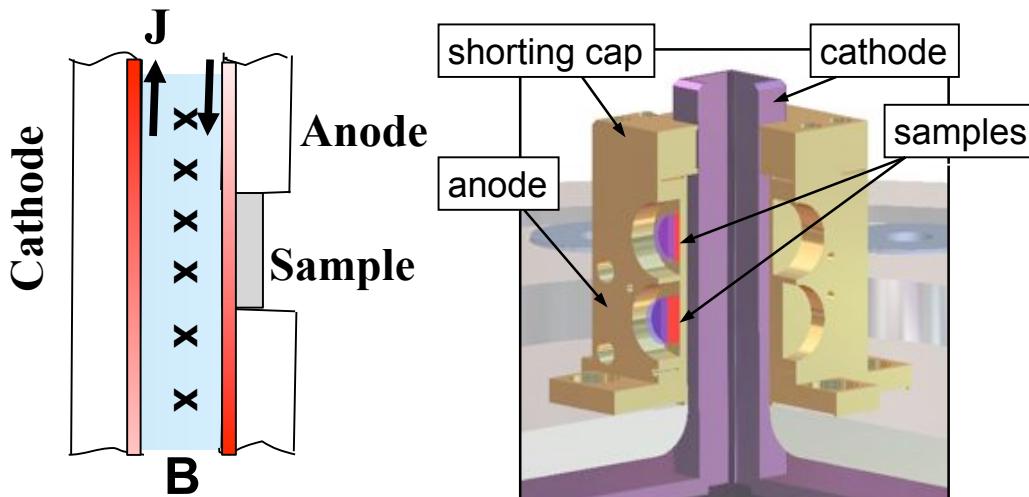
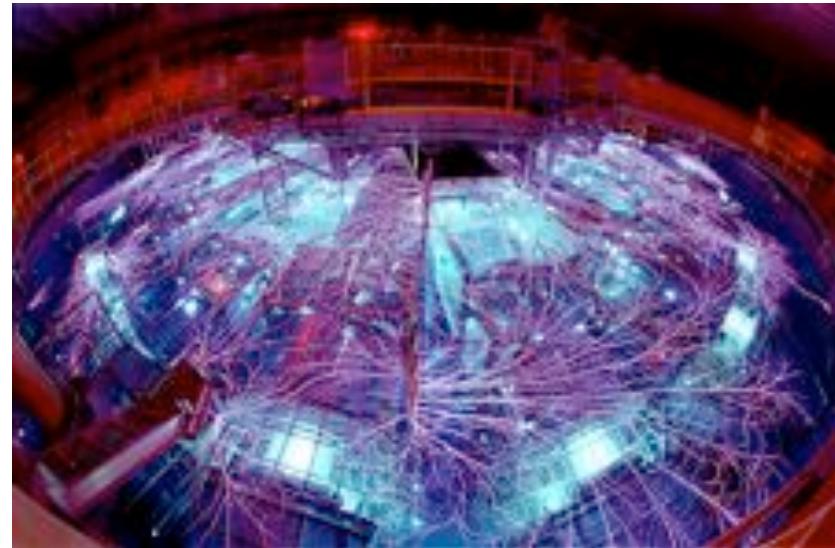
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Z Pulsed Power Machine

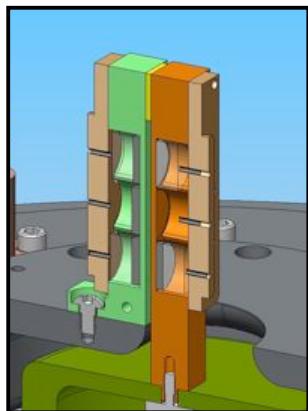
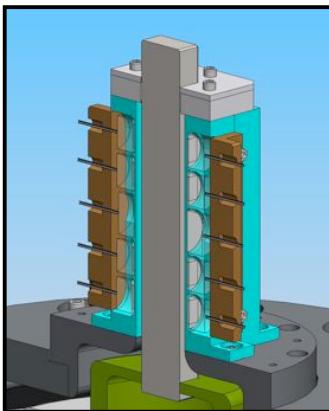
- Designed for ICF applications
- Generates ~26 MA over 100's of ns
- Utilize current to generate magnetic forces
- Magnetic forces create smooth waves in materials
- Waves used for isentropic loading (to ~400 GPa) and to launch high-velocity flyer plates (to ~40 km/s)



Davis, J.-P., Deeney, C., Knudson, M. D., Lemke, R. L., Pointon, T. D., and Bliss, D. E. (2005). "Magnetically driven isentropic compression to multimegabar pressures using shaped current pulses on the Z accelerator," *Physics of Plasmas* 12, 056310.



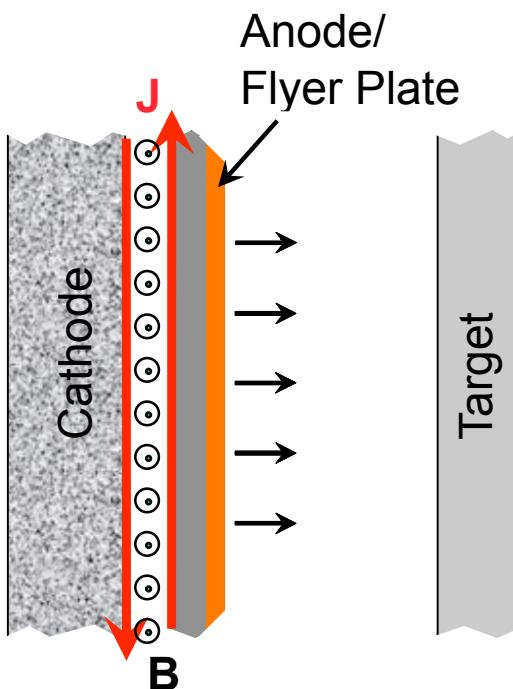
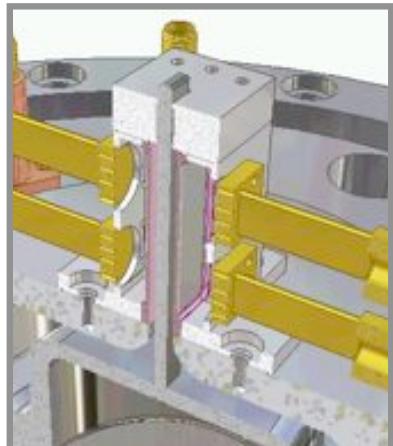
Two Approaches Developed for Accurate EOS Studies – Both Major Advances



Isentropic Compression Experiments (ICE)*

Magnetically driven Isentropic Compression Experiments (ICE) to provide measurement of continuous compression curves to ~4 Mbar
- previously unavailable at Mbar pressures

* Developed with LLNL



Magnetically launched flyer plates

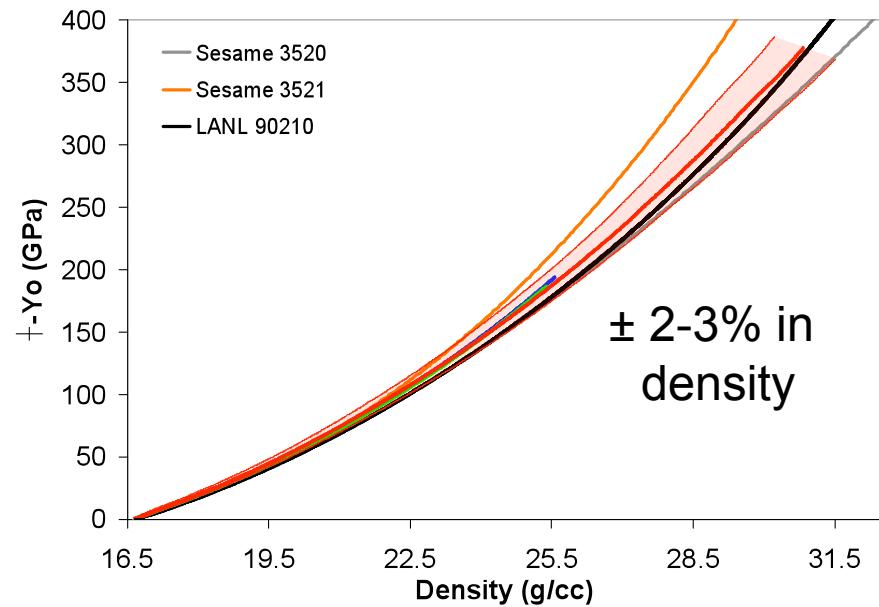
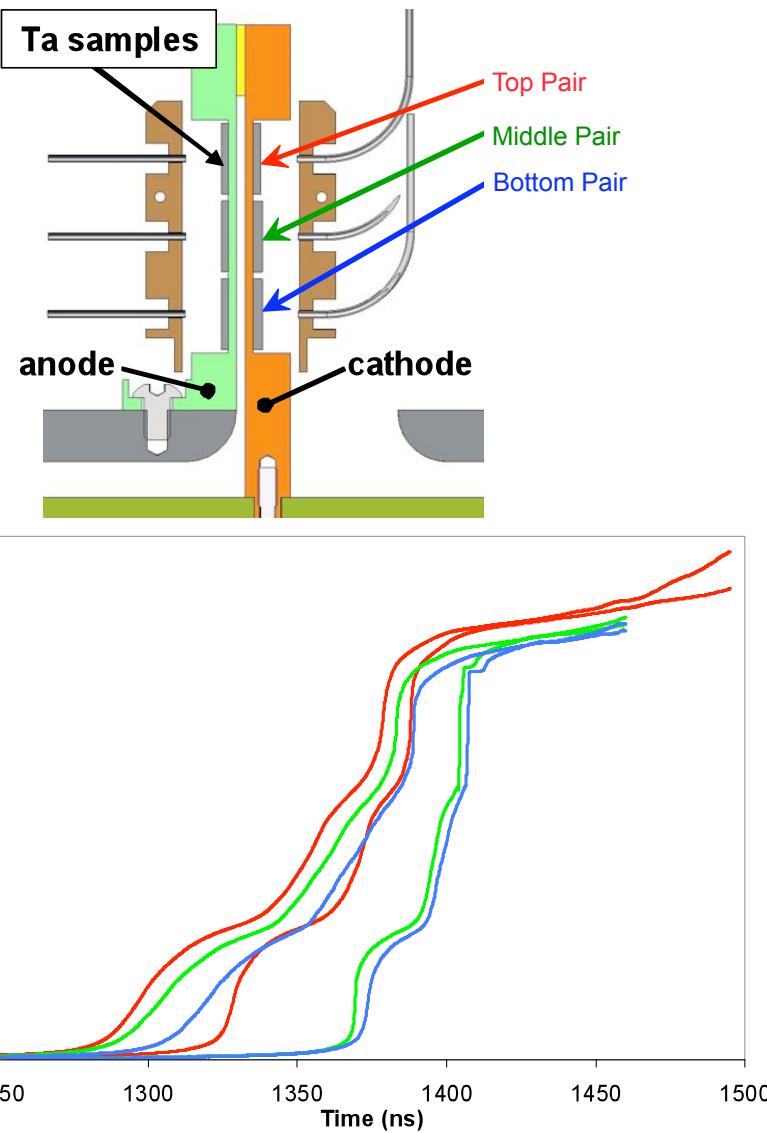
Magnetically driven flyer plates for shock experiments at velocities to > 40 km/s
exceeds gas gun velocities by > 5X and pressures by > 10X with comparable accuracy

Lemke, et al., J. Appl. Phys. 98, 073530 (2005)



Quasi-Isentrope of Ta to nearly 400 GPa

J.-P. Davis

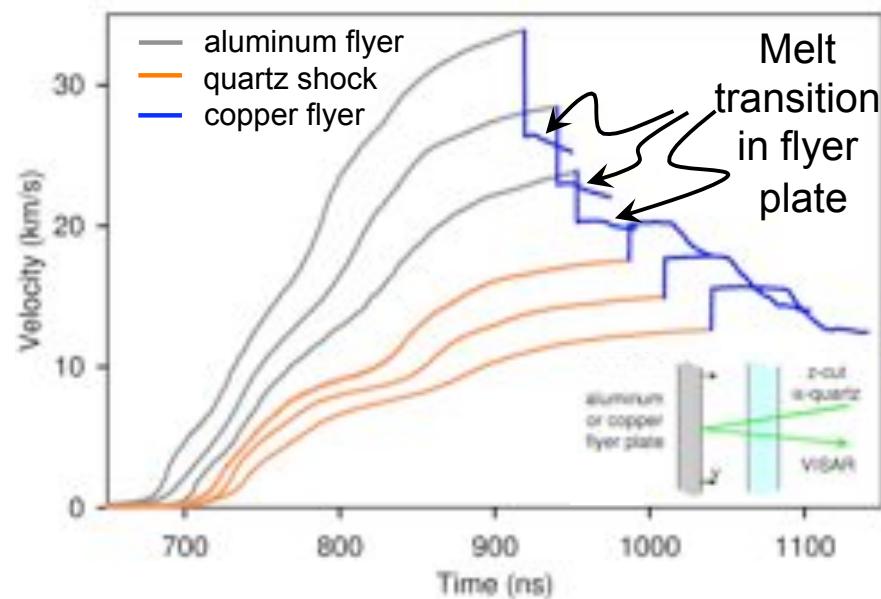
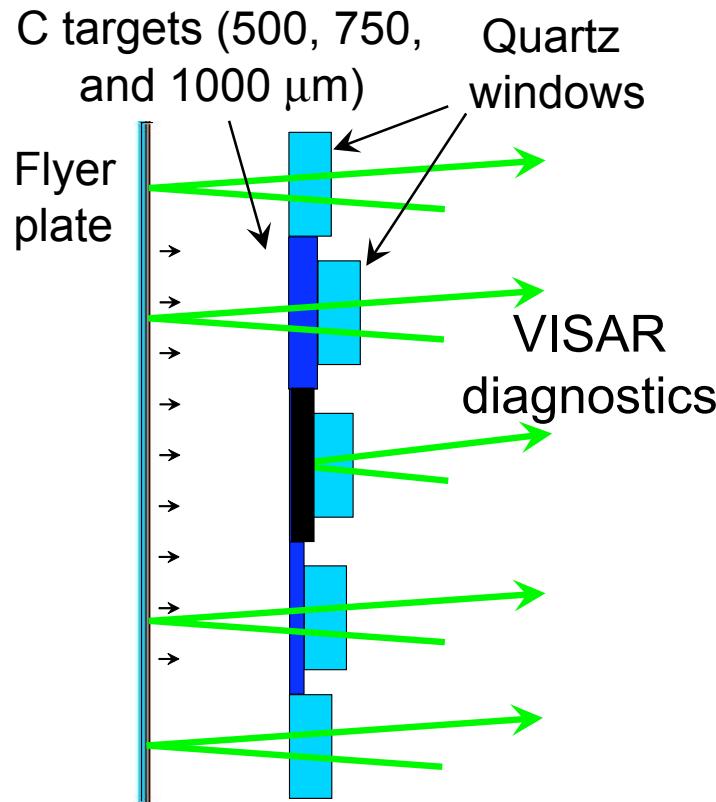


The extracted isentrope
discriminates between various
tabular equations of state for Ta



Quartz of Interest as Standard - Window in Z Shots

M. Knudson
M. Desjarlais



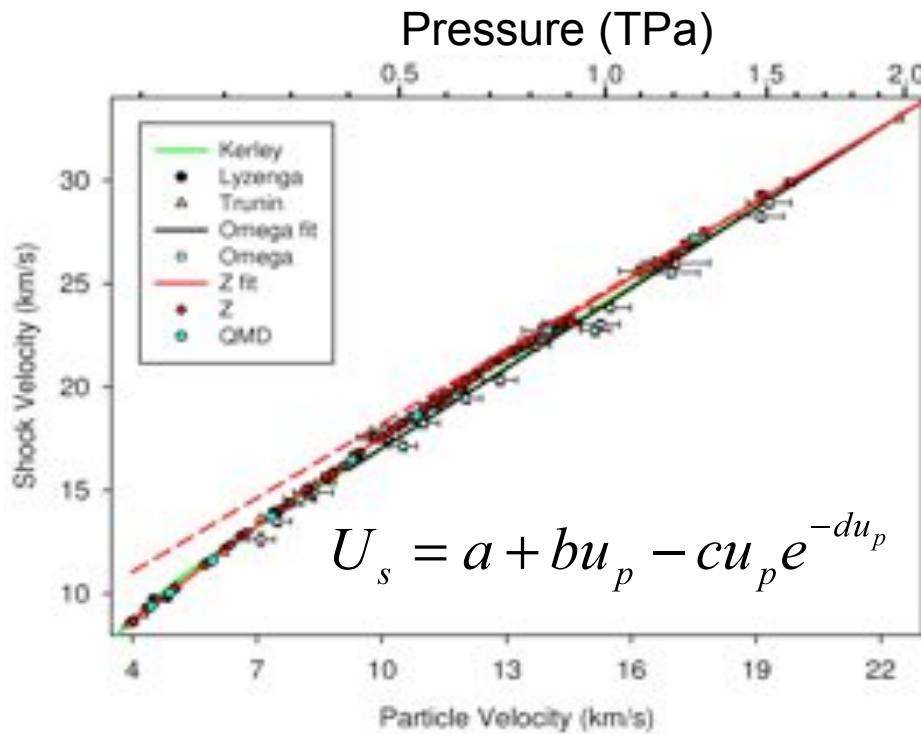
Knudson, M. D., M. P. Desjarlais and D. H. Dolan (2008). "Shock-wave exploration of the high-pressure phases of carbon." *Science 322: 1822-1825.*

quartz data has large uncertainty and scatter, and $U_s - U_p$ exhibits significant curvature in the several 100 GPa regime; attributed to dissociation

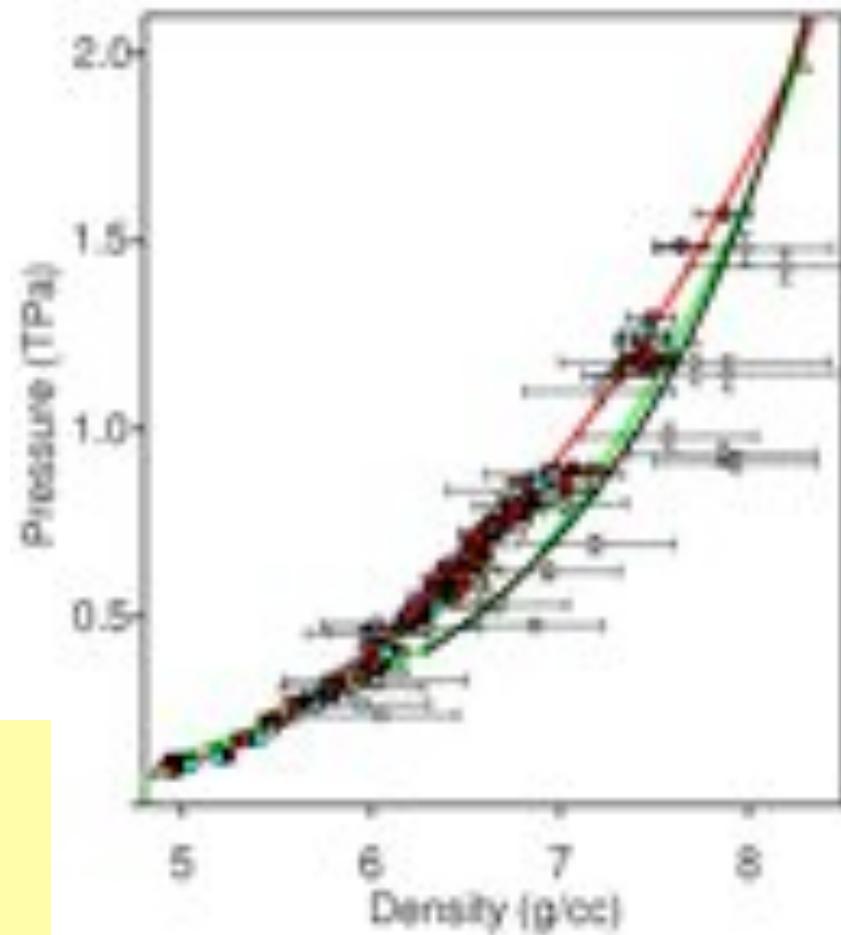


New Hugoniot Data for α -Quartz to 1.5 TPa

M. Knudson
M. Desjarlais



Data set includes ~150 points

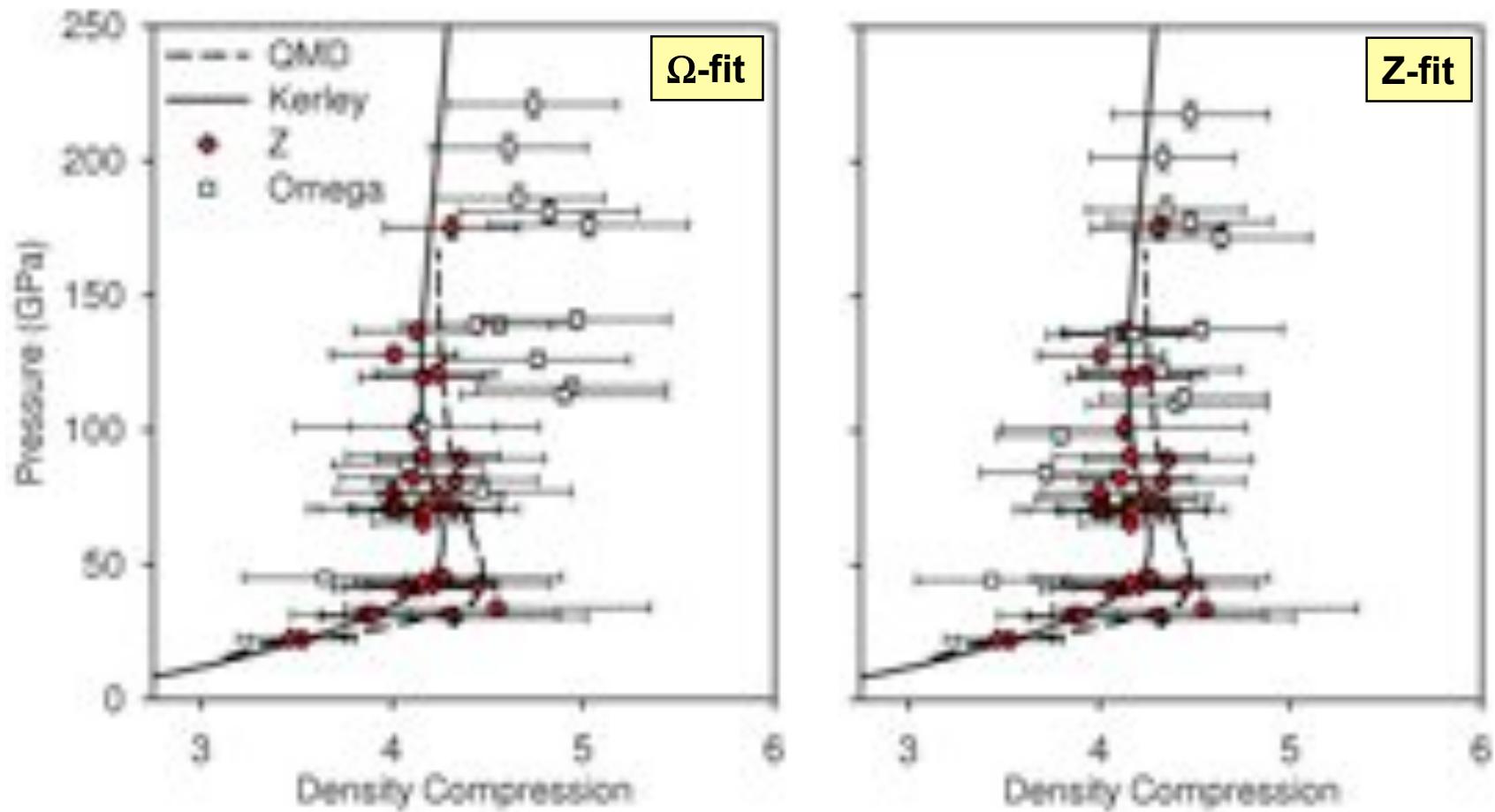


Knudson, M. D. and M. P. Desjarlais
(2009). "Shock compression of quartz to
1.6 TPa: redefining a pressure standard."
Physical Review Letters **103**: 225501.



Deuterium Results Affected by New Quartz Fit

M. Knudson
M. Desjarlais



M.D. Knudson, et al., Phys. Rev. Lett. **87**, 225501 (2001); Phys. Rev. Lett. **90**, 035505 (2003); Hicks, et al., PRB 79, 014112 (2009)

Knudson & Desjarlais, PRL 103, 225501 (2009)



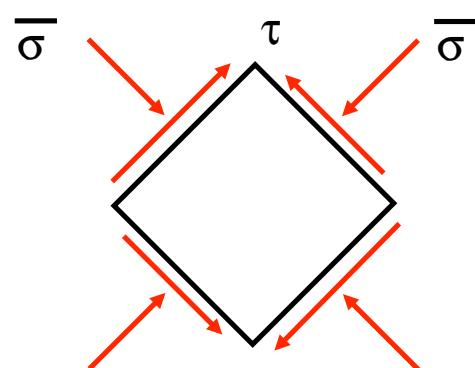
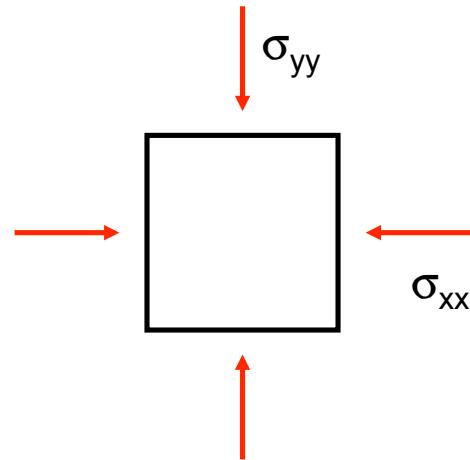
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What is Strength?

- strength is the ability of a material to sustain $\tau_{ij} \neq 0$ or $\sigma_{xx} \neq \sigma_{yy}$
- for a 1-D shock or isentropic experiment, $\sigma_{yy} = \sigma_{zz} \neq \sigma_{xx}$
- conservation equations provide no information about σ_{yy}



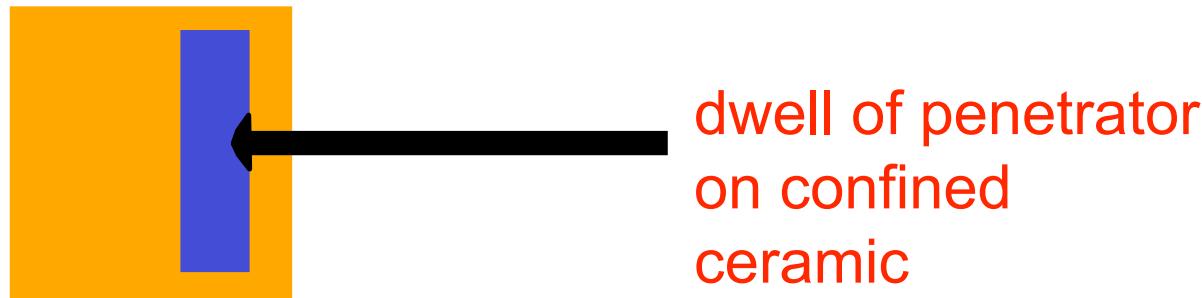
By a simple tensor transformation (45° rotation), the stress state can be expressed as a mean stress (mechanical pressure) and a shear stress

By analogy with uniaxial tension,
 $Y=2\tau_c$



Why Do We Care About Strength?

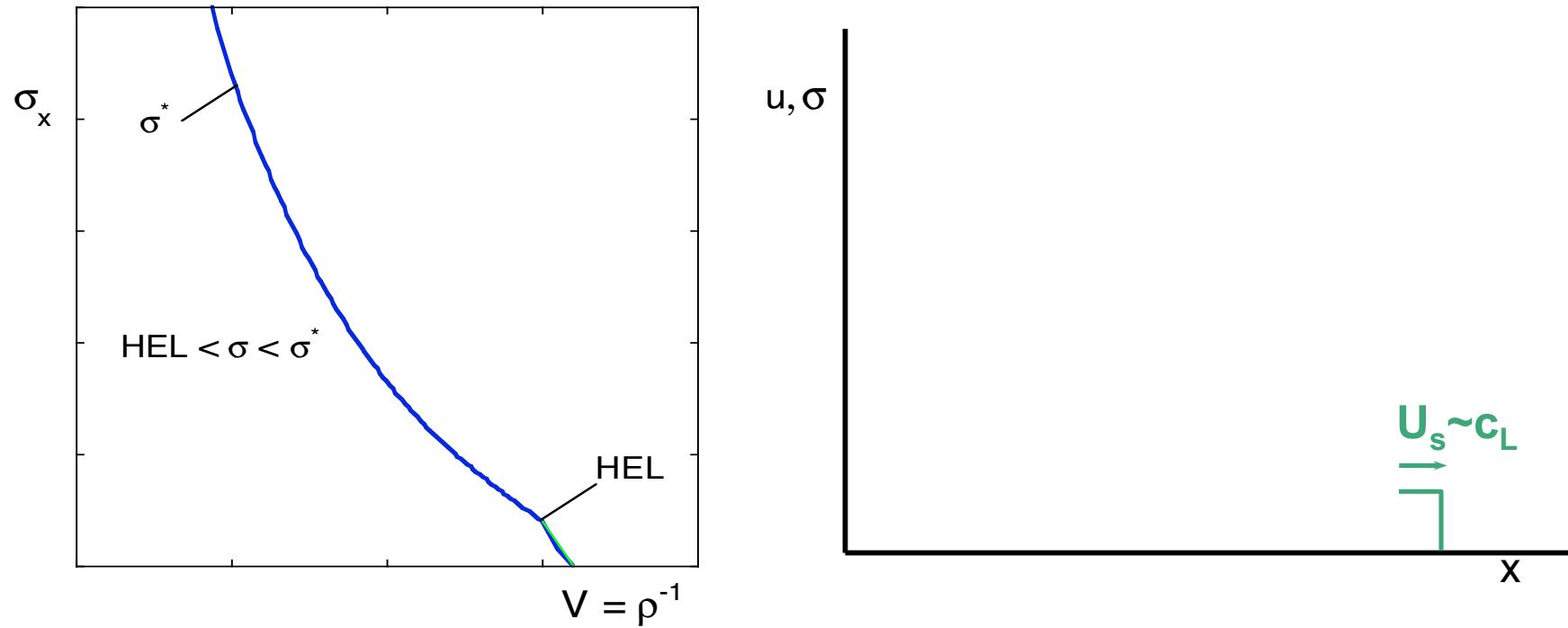
- increases uncertainties in EOS models
- strength affects stress state in diamond anvil cell
- understanding of strength needed for accurate computational results
- Rayleigh-Taylor instabilities inhibited by strength
- weapons and armor applications (ceramic armor, penetrators, etc.) influenced by strength



dwell of penetrator
on confined
ceramic



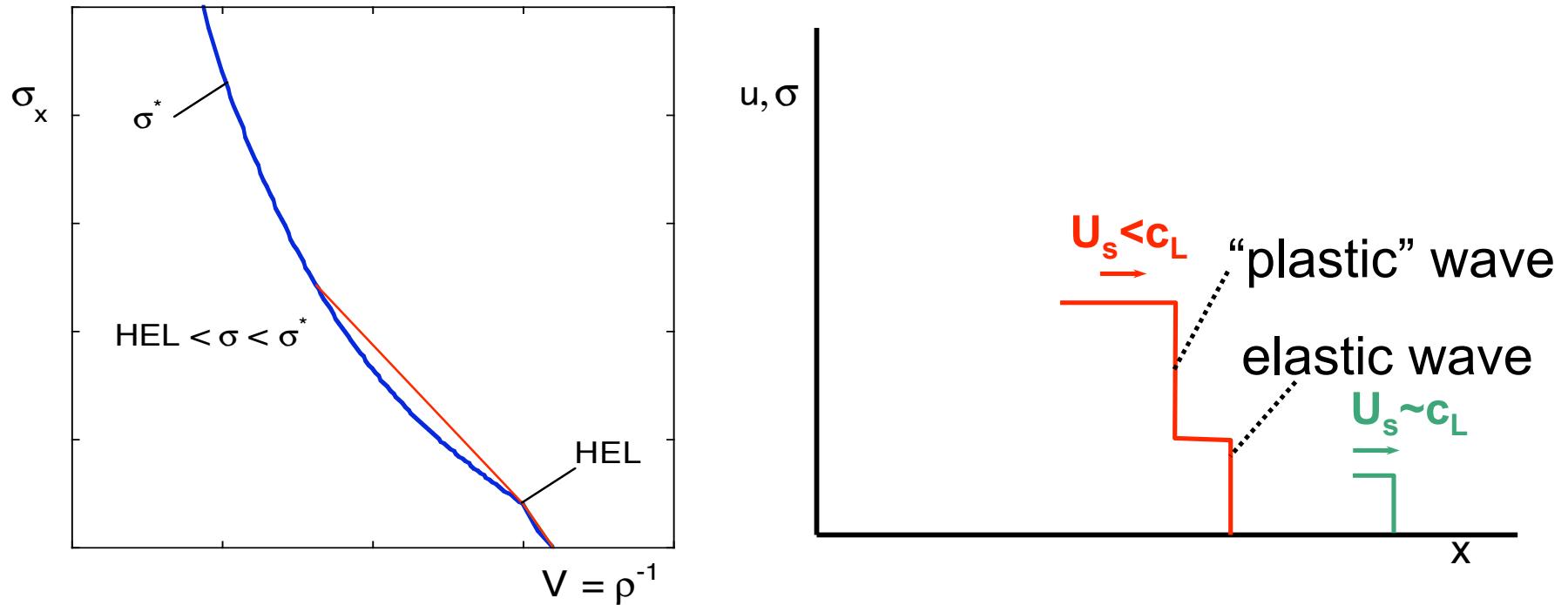
Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses



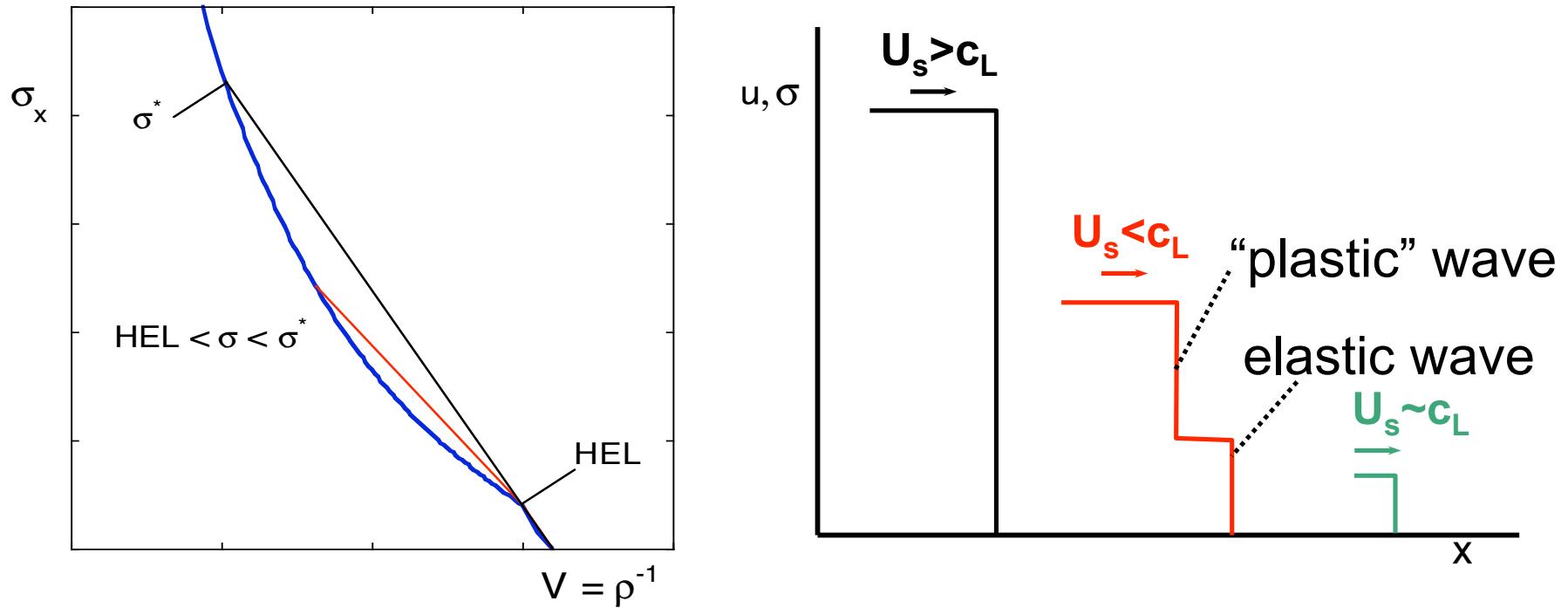
Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses
- for stresses somewhat above HEL (Hugoniot elastic limit), a two-wave structure develops



Effect of Strength on Shock Waves



- most solids display elastic behavior for low shock stresses
- for stresses somewhat above HEL (Hugoniot elastic limit), a two-wave structure develops
- for high stresses, plastic wave travels faster than elastic precursor

above HEL, σ_y is unknown regardless of stress level!



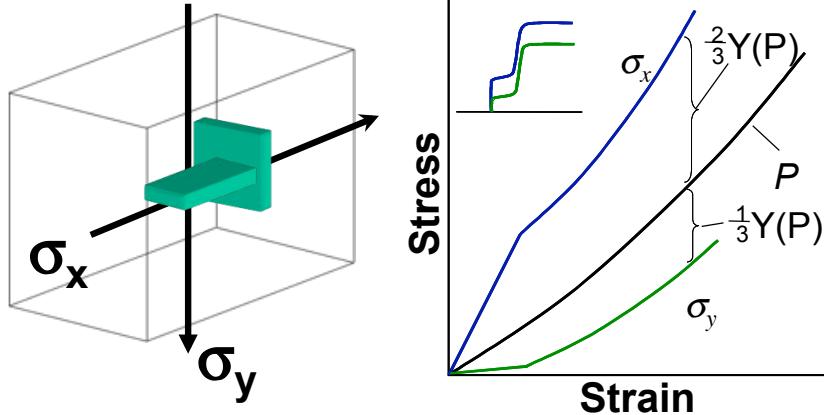
Typical Values of HEL

Material	HEL (GPa)
6061-T6 Al	0.5
tungsten	3 - 4.5
silicon carbide (SiC)	12 - 14
boron carbide (B ₄ C)	15 - 18
diamond	60 - 100

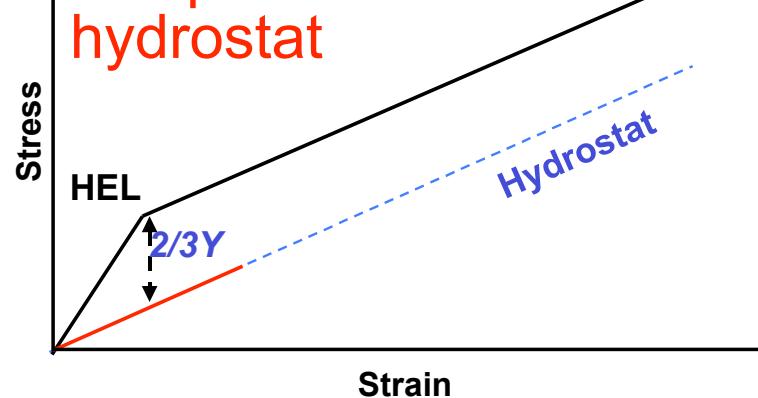


Techniques to Determine Strength at High Pressures

lateral gauges



comparison to hydrostat



- *stress gauges can provide independent measures of σ_x and σ_y to determine dynamic strength*
- *calibration of gauges difficult*
- *only function to ~ 20 GPa due to shorting of insulation*

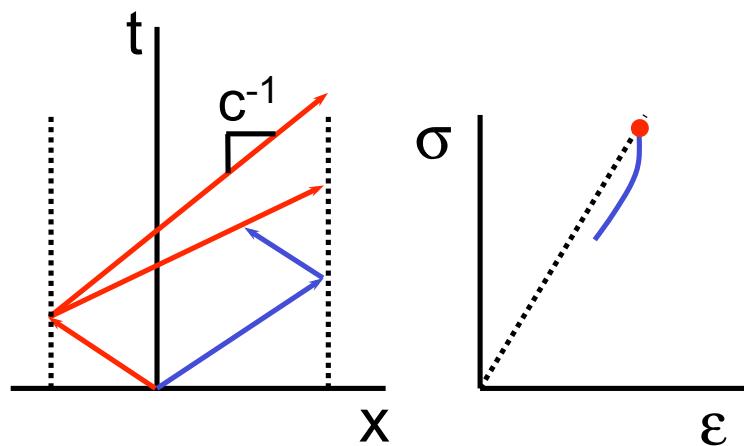
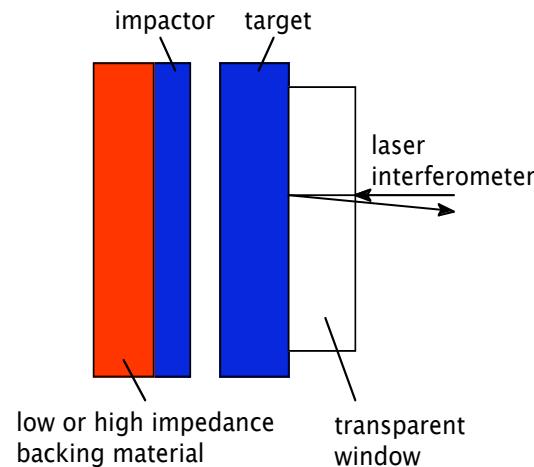
- *most common method: compare with hydrostatic data*
- *hydrostat from ultrasonic sound speed data (C vs. P) or diamond anvil cell (P - V)*
- *uncertainties can be very large*

also: X-ray diffraction, pressure-shear loading, growth of Rayleigh-Taylor instabilities, within diamond anvil cells

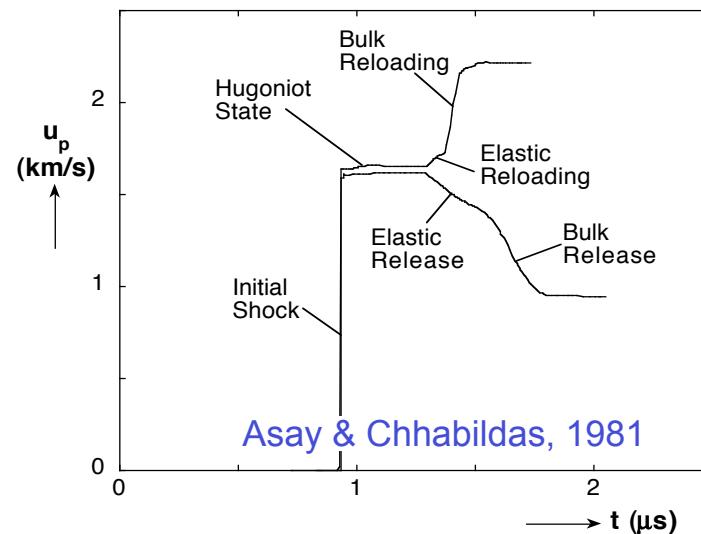


Self Consistent Method (Asay, Lipkin, Chhabildas, et al.)

reshock & release
configuration



results for 6061-T6 Al



- wave speed determined from VISAR release or reloading profiles
- unloading path calculated from incremental relations

strength based on relative difference from Hugoniot state



Past Work: the Self Consistent Method

Sandia - reshock/release of Al, Be, Cu, Mo, Ta, V, W
(Asay, Chhabildas, *et al.*)

WSU - reshock/release of Al

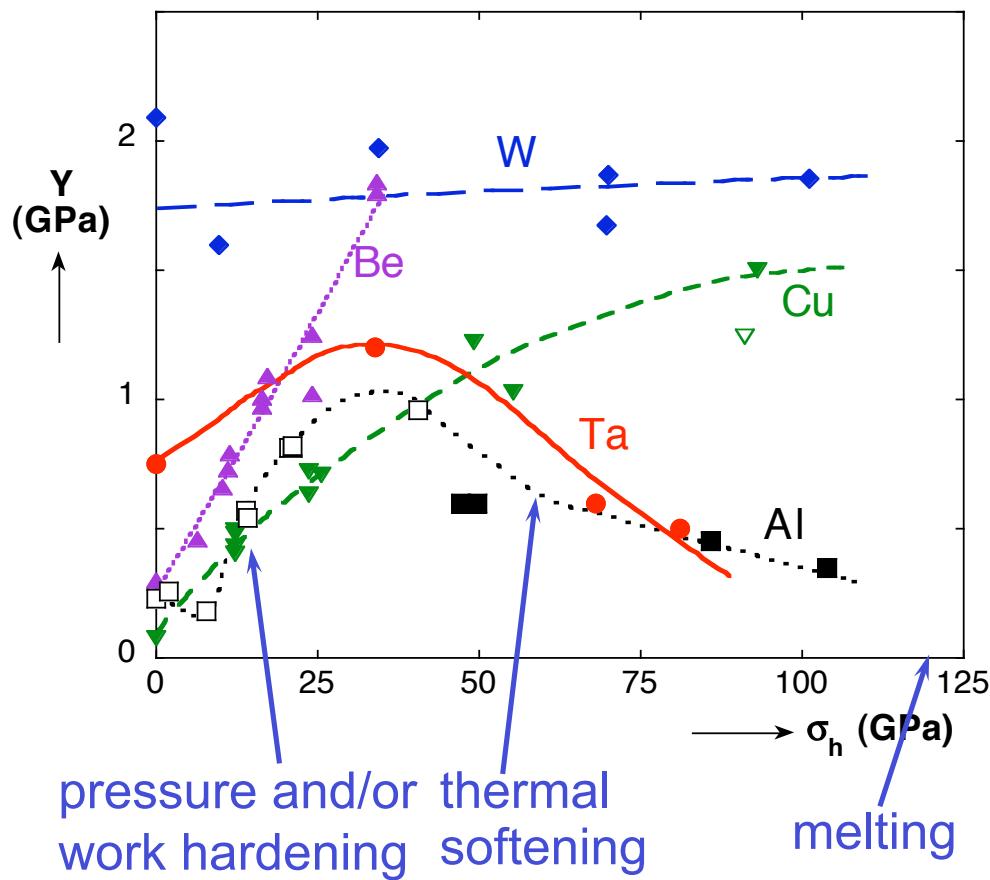
LANL - release of 2024 Al

Russia - results of an experiment by Al'tshuler *et al.*, 1999, using longitudinal gauges agree with results of Asay & Chhabildas [1981] and with lateral gauge measurements



Strength of Metals Under Shock Loading

metals studied to high pressure using self-consistent method



- strength increases due to work and pressure hardening
- ultimately decreases due to thermal softening



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Strength of Armor Ceramics

- Al_2O_3 , B_4C , and SiC considered or fielded for personnel or vehicle protection
- ceramic armor performance closely linked to specific configuration --> need for accurate models for use in armor design
- “conventional wisdom” says (most) ceramics lose strength when shocked



Reinhart, W. D. and L. C. Chhabildas (2003). "Strength properties of Coors AD995 alumina in the shocked state." *International Journal of Impact Engineering* **29**: 601-619.

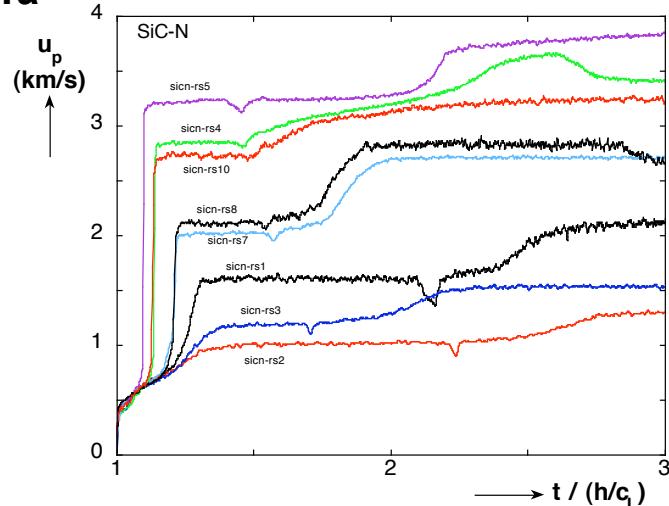
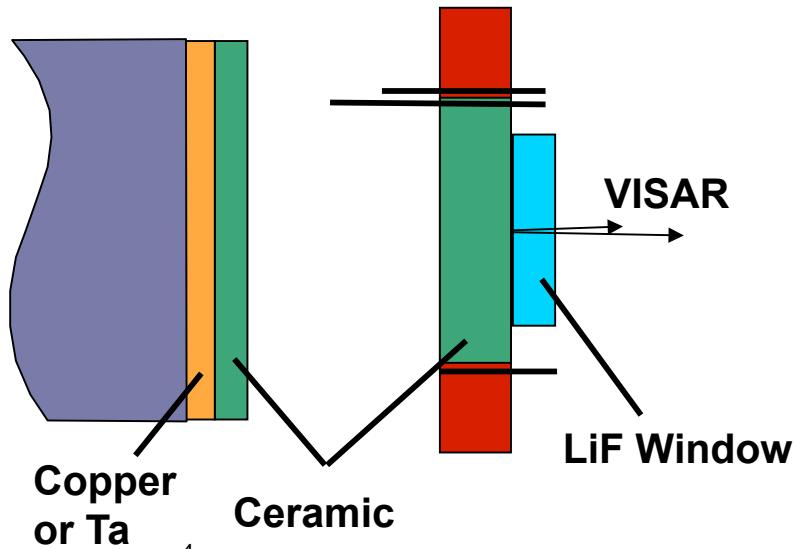
Vogler, T. J., W. D. Reinhart and L. C. Chhabildas (2004). "Dynamic behavior of boron carbide." *Journal of Applied Physics* **95**: 4173-4183.

Vogler, T. J., W. D. Reinhart, L. C. Chhabildas and D. P. Dandekar (2006). "Hugoniot and strength behavior of silicon carbide." *Journal of Applied Physics* **99**: 023512.

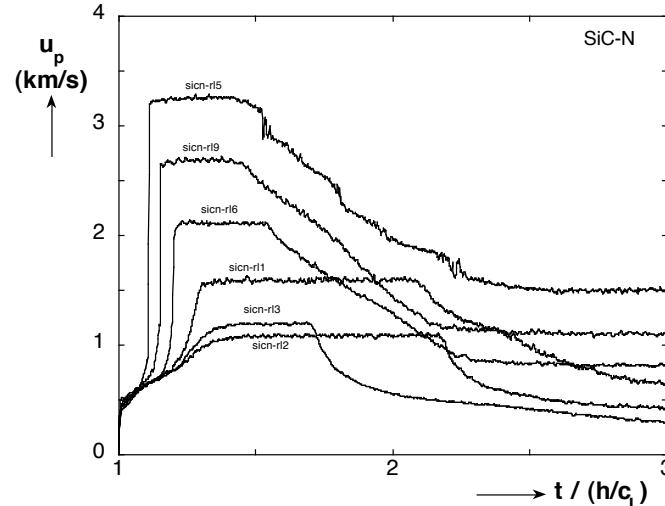
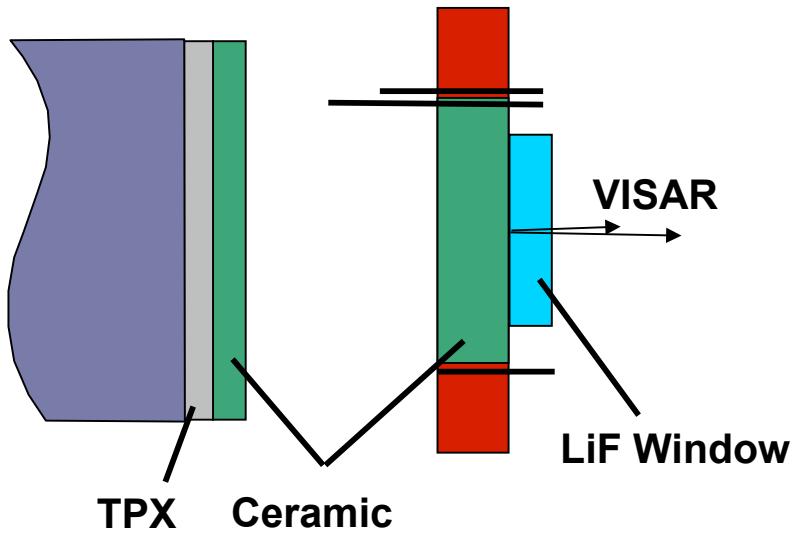


Reshock & Release Experiments for B_4C and SiC

Reshock Configuration

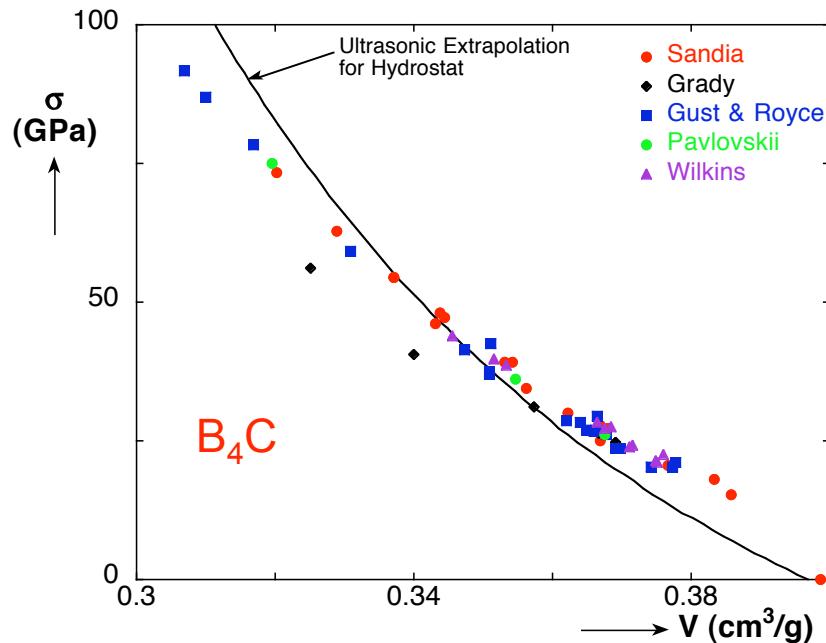
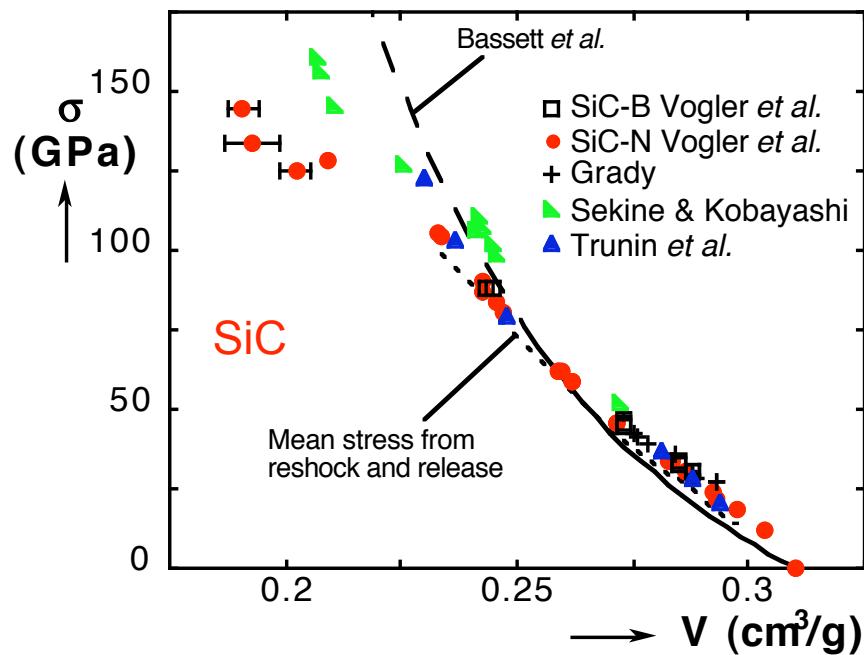


Release Configuration





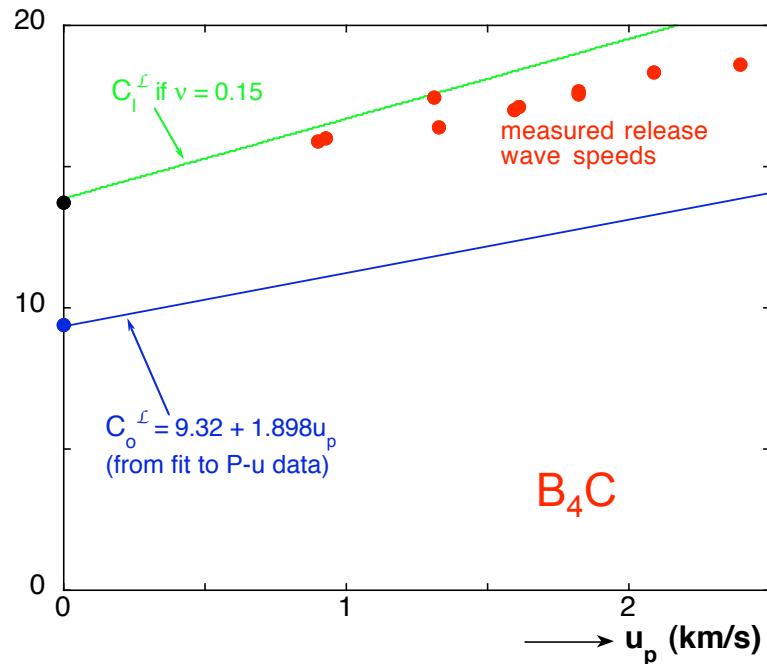
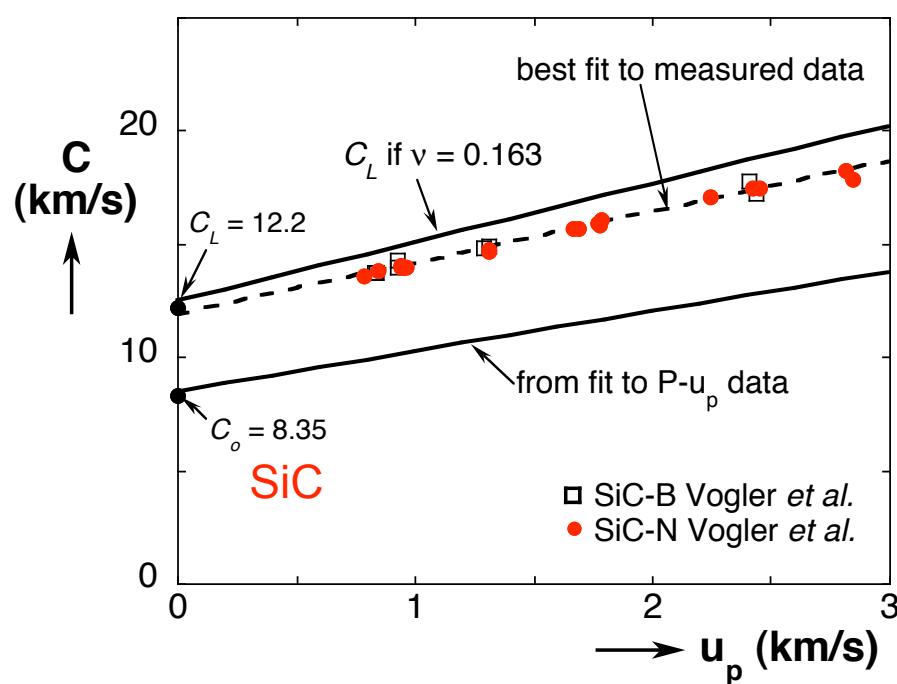
Hugoniots for SiC and B₄C



- Sandia studies have completed data sets, agreement with other studies generally good
- phase transition observed in SiC at ~ 105 GPa
- possible phase transition in B₄C at ~ 40 GPa, but evidence inconclusive (behavior may be due to shock amorphization)



High Release Wave Speeds Are Indicative of Elastic Behavior



- release waves arrive much faster than expected bulk wave speeds (based on Hugoniot)
- v appears to be increasing in B_4C

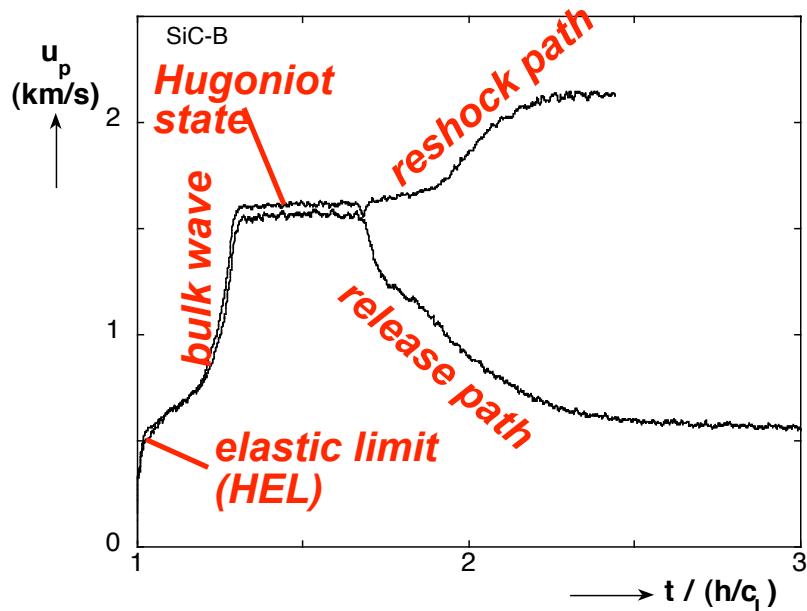
$$C_{bulk} \approx \frac{1}{\rho_o} \frac{d\sigma}{du} \Big|_{Hugoniot}$$

material displays elastic behavior in shocked state



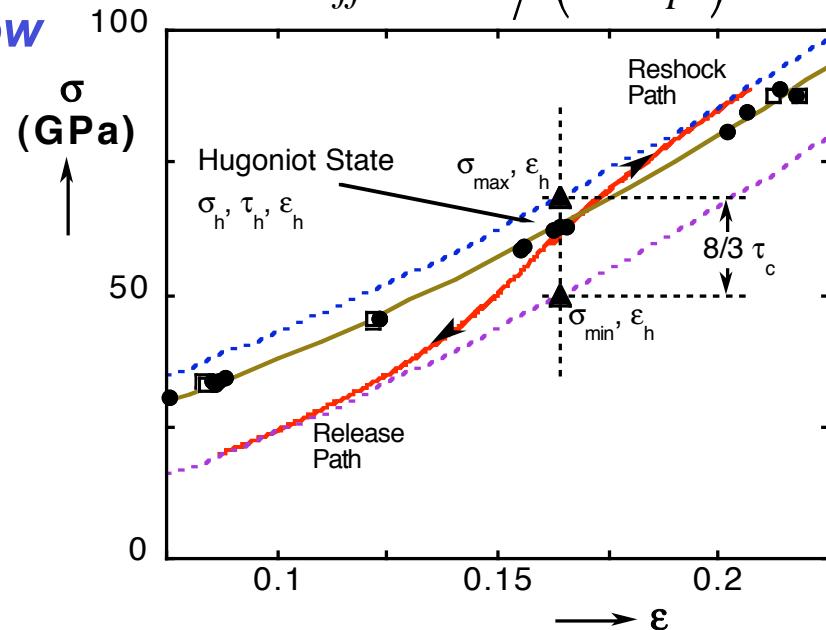
Analysis of Reshock & Release Experiments

- *Incremental form of conservation equations used to calculate σ - ϵ paths*
$$\Delta\sigma = \rho_o c \Delta u$$
$$\Delta\epsilon = \Delta u/c$$
- *Account for presence of window*



- *most of reshock path arrives later than release*
- *both arrive at elastic wave speed*

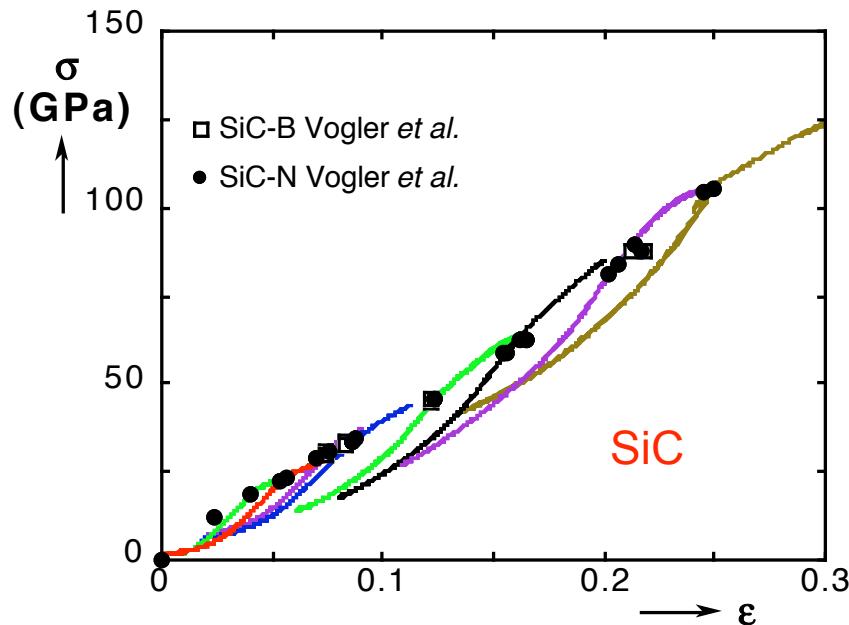
$$c \text{ based on effective shock: } U_{eff} = \sigma^H / (\rho_o u_p^H)$$



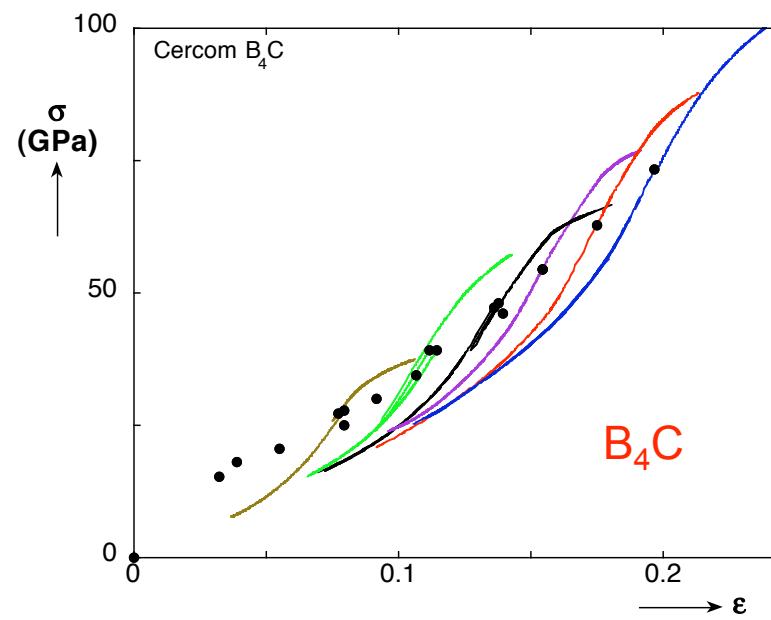
- *Technique to estimate:*
 - *Strength of the material in the shocked (damaged) state*
 - *Dynamic mean stress curve*
 - *Shear stress in Hugoniot state*



Reshock & Release Paths



- release paths below Hugoniot
- reshock paths only slightly above Hugoniot
- shallow reshock >100 GPa due to phase transformation

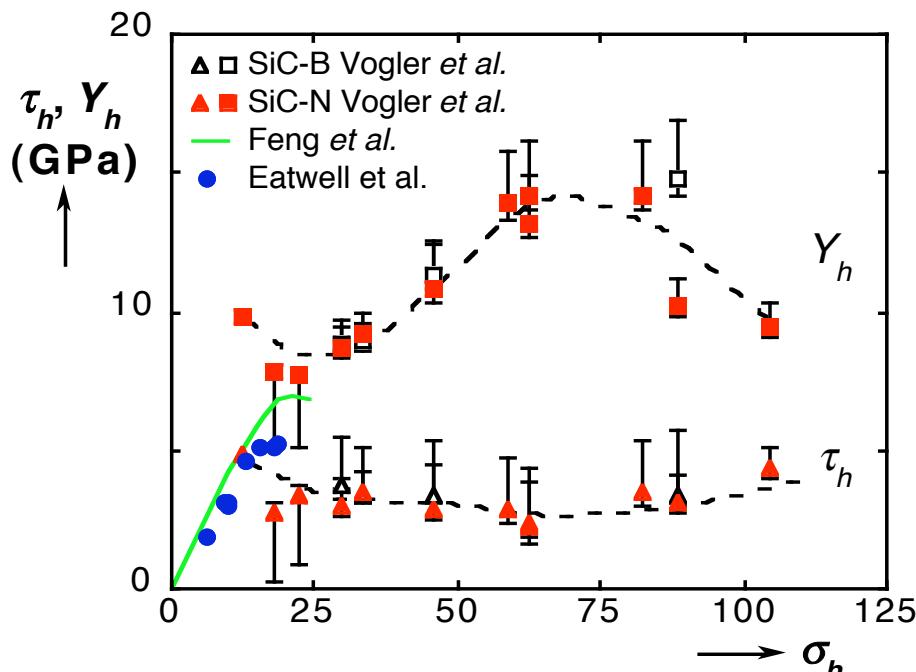


- release paths below Hugoniot
- reshock paths significantly above Hugoniot

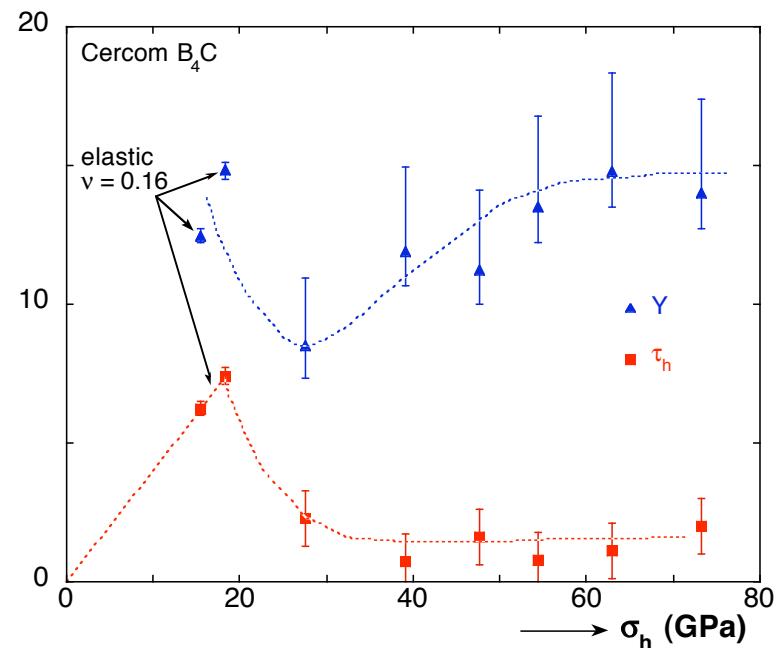
comparable behavior on release; SiC behaves in “metal-like” manner while B_4C behaves unexpectedly



Strength in Shocked State



- significant strength retained
- apparent loss of strength approaching phase transition
- τ_h remains approximately constant



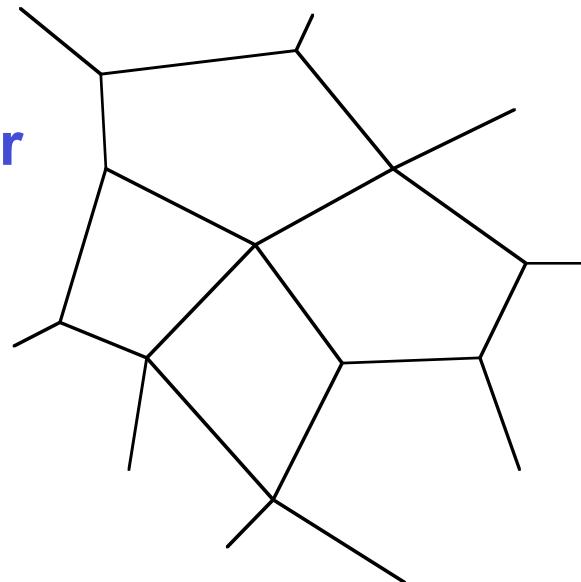
- material retains only small shear stress in Hugoniot state but large apparent strength during reshock/release
- strongly suggestive of damage
- kinetics and physical mechanisms unclear

how important is this behavior in an armor application?



Behavior of B_4C (and Al_2O_3): A Possible Physical Explanation

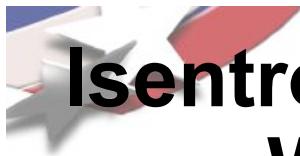
- initial shock creates damaged (micro-cracked) material and possible local melting but relatively small relative motion
- material equilibrates after shock passage; melted material solidifies; interfaces at rest or “healed” by solidification (similar to concept of Swegle & Grady, 1986)
- when perturbed by release or reshock, material responds elastically



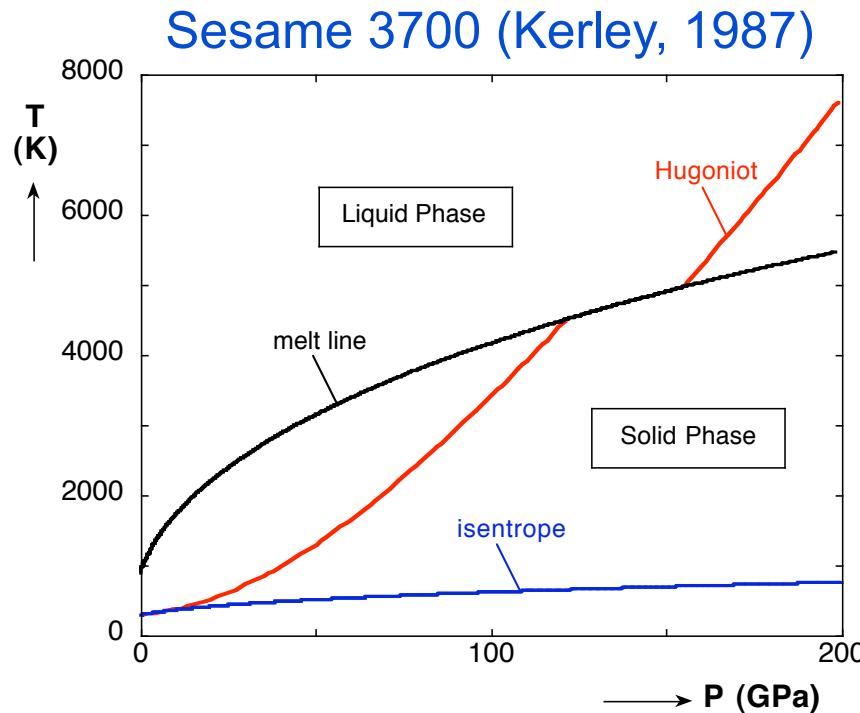


Outline of Talk

- Introduction to shock and high-pressure physics
- Gas gun technology and examples
- The Z machine and examples
- Background on high pressure strength and techniques to measure it
- Strength of ceramics under shock loading
- Strength of metals under isentropic loading
- Conclusions



Isentropic Loading Accesses Cool Regimes Where Strength Is More Important



Steinberg-Guinan Strength Model (rate-independent version):

$$\frac{Y}{Y_o} = (1 + \beta(\varepsilon_p + \varepsilon_i))^n \frac{G(P, T)}{G_o} \quad G(P, T) = G_o + \frac{\partial G}{\partial P} \frac{P}{\eta^{1/3}} + \frac{\partial G}{\partial T} (T - T_o)$$

strength larger under isentropic loading due to smaller ΔT



Strength Under Quasi-Isentropic Loading: Previous Work

aluminum - isentrope stiffer than Hugoniot to 9 GPa (Barker-SNL)

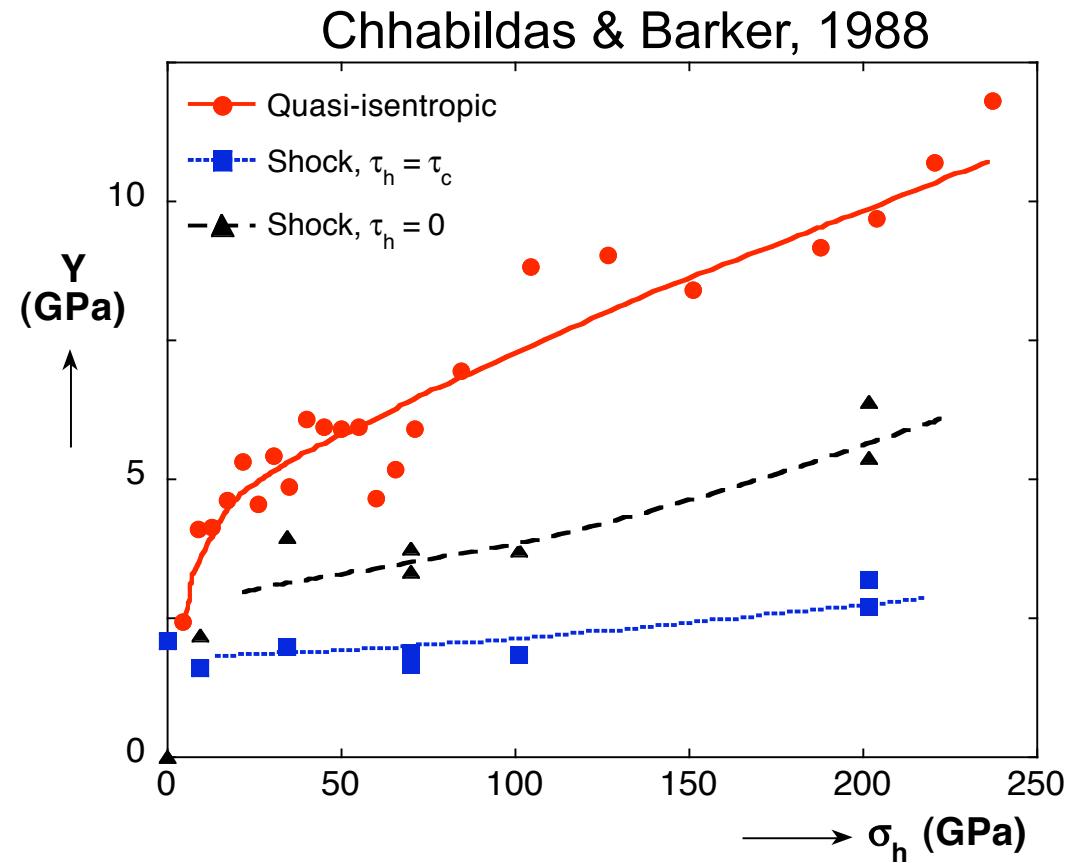
tungsten - isentrope from graded-density impactor lies above Hugoniot up to 140 GPa due to strength

molybdenum - isentrope from Z compared with hydrostat to estimate strength (Reisman-LLNL/SNL)

lateral gauges

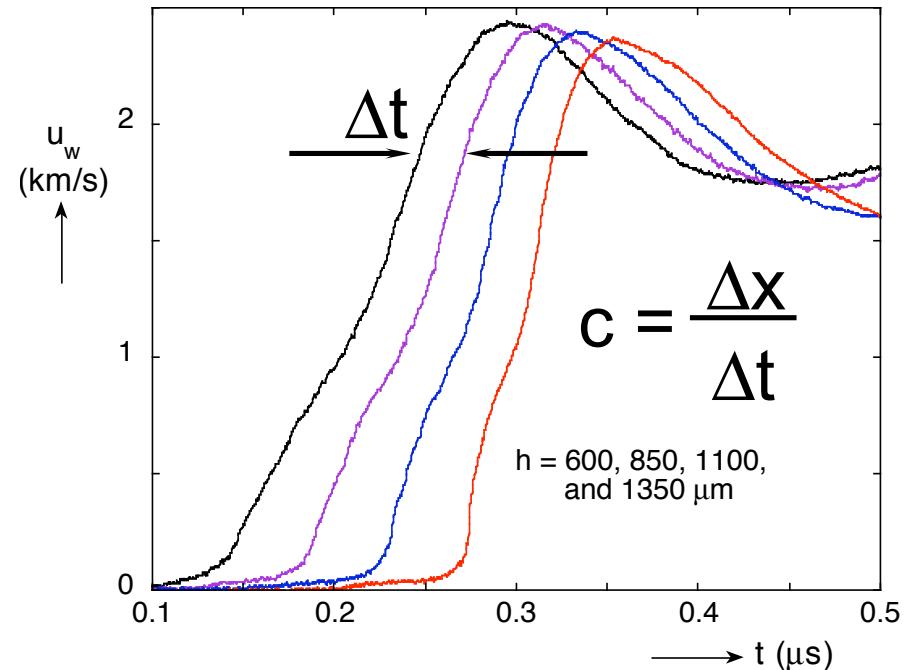
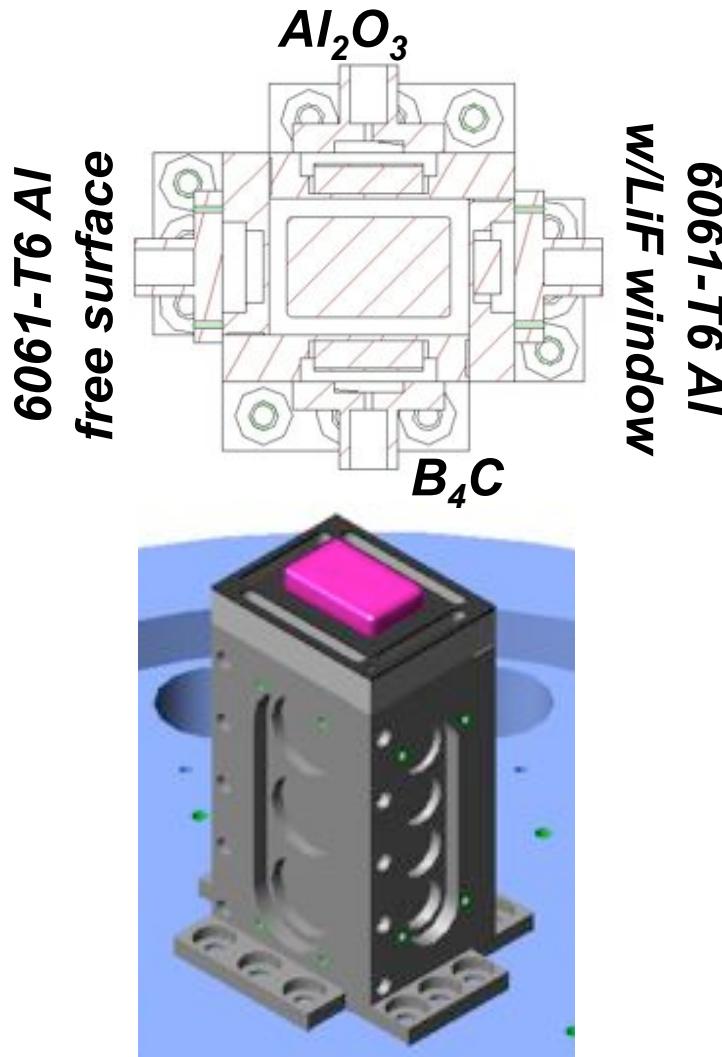
- AD-1 aluminum and copper (Bat'kov *et al.*)
- copper, iron, steel (Rosenberg *et al.*)

growth of Rayleigh-Taylor instabilities - aluminum and vanadium (LLNL)





Results for 6061-T6 Aluminum to 55 GPa



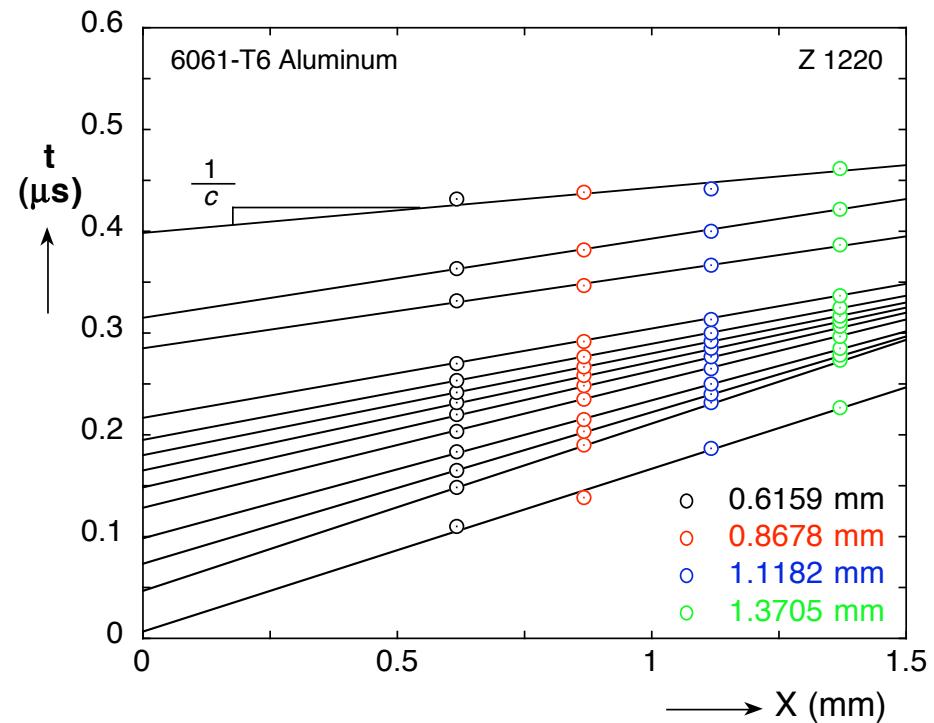
- four profiles, both loading and unloading histories
- peak attenuation evident
- unloading structure develops in thicker samples



Lagrangian Analysis Technique

- *backwards integration technique of Hayes is non-unique for elastic-plastic materials*
- *Lagrangian analysis technique follows previous work by Grady and others*

1) determine $c(u_w)$ by least-squares fit to VISAR data

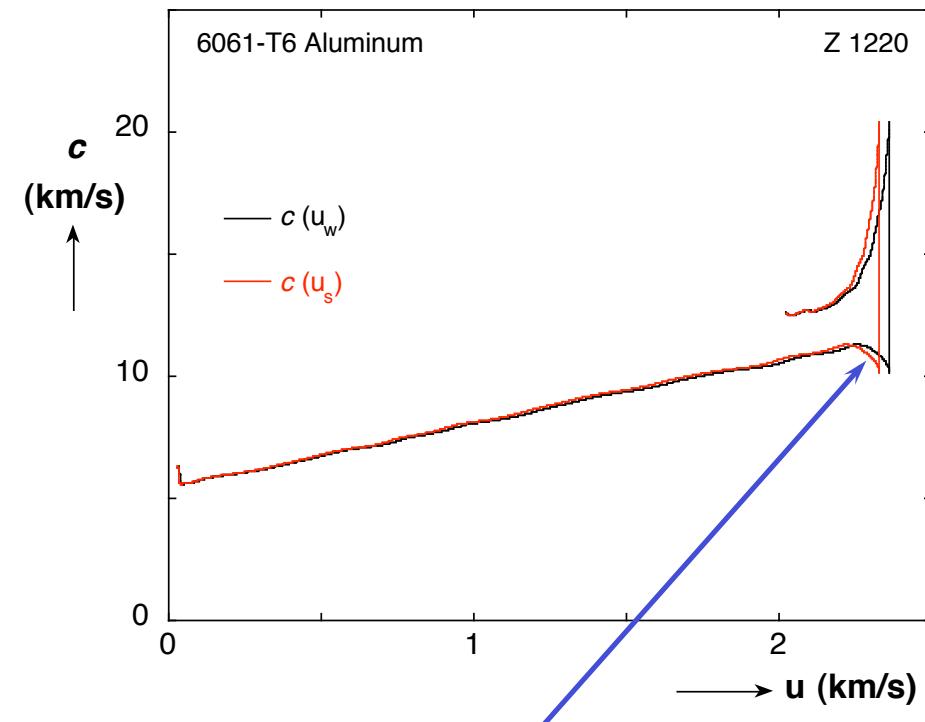




Lagrangian Analysis Technique (2)

2) correct $c(u_w)$ to $c(u_p)$ by impedance matching of window and sample

$$\Delta u_s = \frac{Z_s + Z_w}{2Z_s} \Delta u_w$$



relaxation due to drop in strain rate?



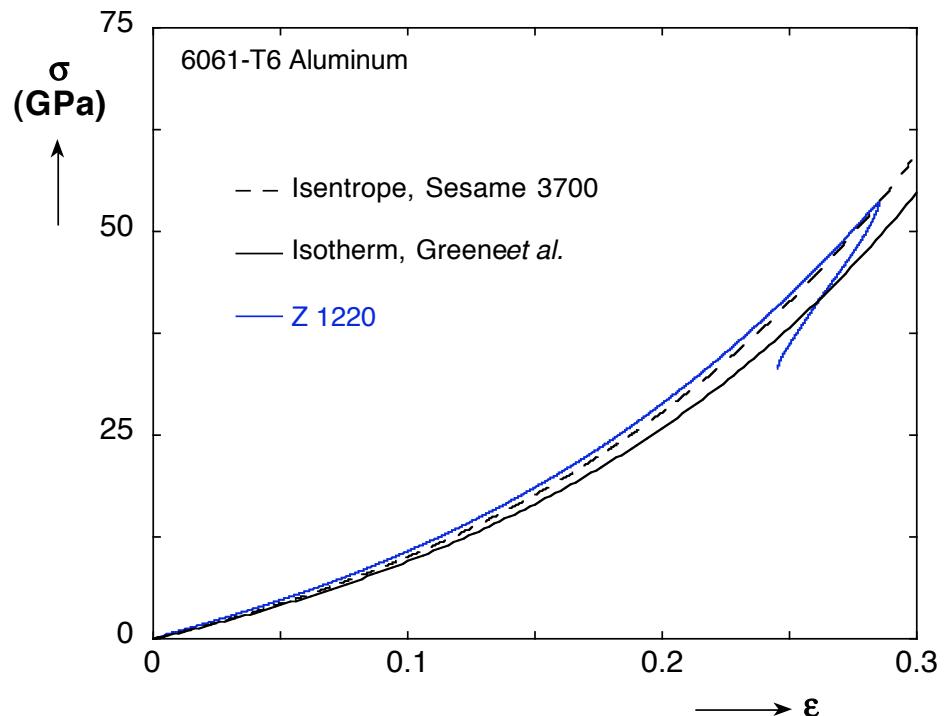
Lagrangian Analysis Technique (3)

3)integrate stress and strain using incremental relations:

$$\Delta\sigma = \rho_o c \Delta u_p$$

$$\Delta\epsilon = \Delta u_p / c$$

for unloading, ignore attenuation by beginning at lowest peak u_p





Assumptions in Lagrangian Analysis Technique

- characteristics not bent by window interactions
- rate-independent material
- all points experience same loading history
- window behavior known
- window loads along its principal isentrope

none of assumptions fully met!

none is too bad, either!

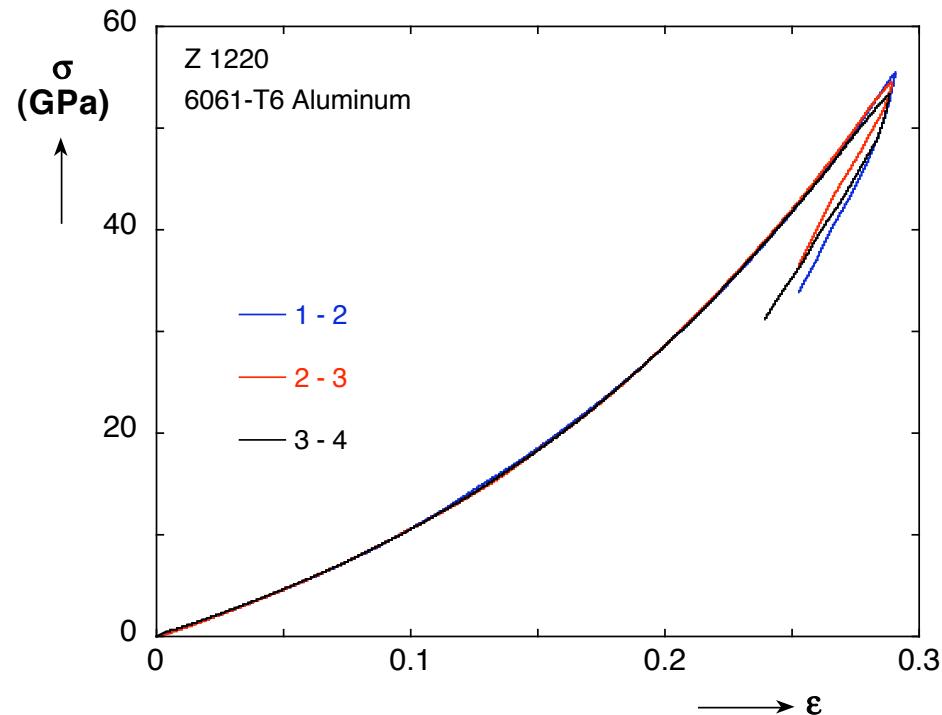


Strength Measurement

Since stress-strain histories are somewhat different for each material point, VISAR results are analyzed in a pairwise fashion.

Loading responses are very similar for the three pairs.

Difference between loading and unloading curves is a measure of the strength.



$$Y = \frac{3}{4}(\sigma_L - \sigma_U)$$

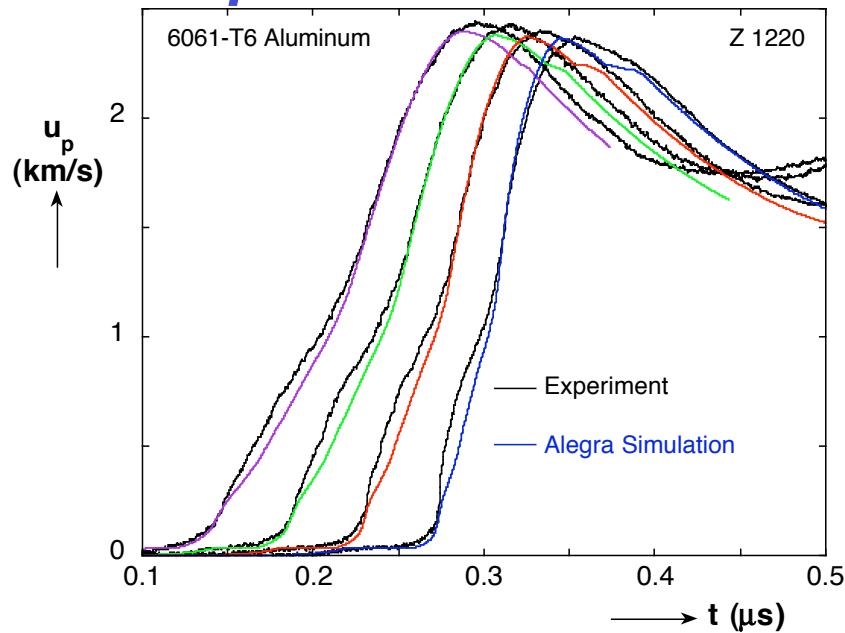
~~$$= 3.1, 2.2, 1.5 \text{ GPa}$$~~

results suspect due to wave interactions



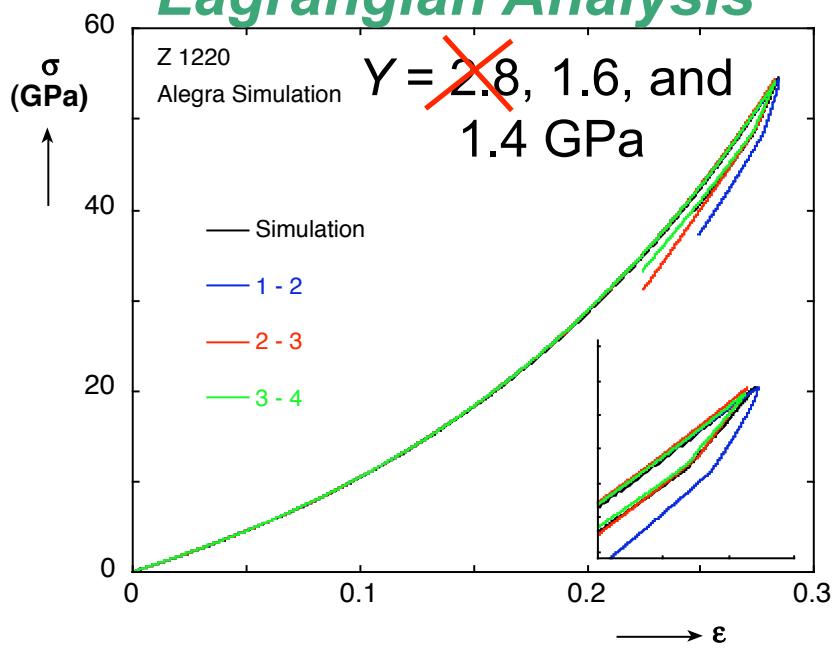
Simulated Data for Z 1220

Experiment & Simulation



- 1-D Alegra simulation - arbitrary Lagrangian-Eulerian magneto-hydrodynamics code, Steinberg-Guinan-Lund strength model
- current input somewhat off on loading, too much structure in unloading wave

Lagrangian Analysis

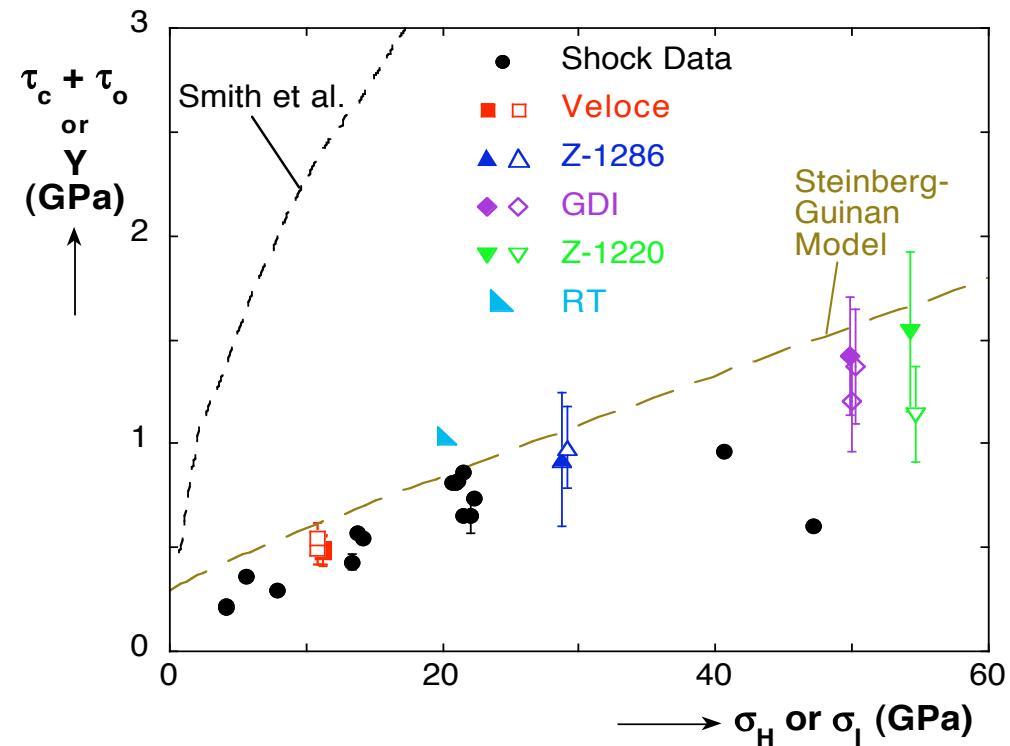


- loading captured extremely well
- first pair affected by window perturbations reaching drive surface, but others accurate measurements of Y (1.48 GPa)



Strength Under Isentropic and Shock Loading

- strength under isentropic loading continues to increase with increased stress (due to pressure and/or work hardening)
- current experimental results agree with Steinberg model
- results of Smith et al. significantly different



Asay, J. R., Ao, T., Davis, J.-P., Hall, C. A., Vogler, T. J., and Gray, G. T. (2008). "Effect of initial properties on the flow strength of aluminum during quasi-isentropic compression," *J. Appl. Phys.* **103**, 083514.

Vogler, T. J., Ao, T., and Asay, J. R. (2009). "High-pressure strength of aluminum under quasi-isentropic loading," *Int. J. Plast.*, **25**, 671-694.



Conclusions

- self-consistent technique can be applied to shock or ramp loading for metals or ceramics
- SiC retains significant strength and shear stress in shocked state in a “metal-like” manner
- B_4C
 - retains considerable strength in the shocked state
 - shear stress is low or zero
 - current concepts of strength inadequate to describe this behavior
 - physical mechanisms unclear - damage or “thermal trapping”?
- strength measurements under isentropic loading to 60 GPa for aluminum
- strength results are relatively insensitive to timing errors (1 ns shift gives ~8% error in Y at 60 GPa)
- conservative estimate of 20-30% uncertainty on Y at 50 GPa for Al,



Work Needed

- comparison of different techniques (e.g. Rayleigh-Taylor, DAC, and self-consistent) for same materials
- utilize pressure-shear techniques to decouple effects of pressure, strain rate, work hardening, etc.
- grain scale investigations to understand mechanisms associated with strength retention/loss in metals and ceramics
- establish connections between experiments and molecular dynamics simulations
- investigate effect of solid-solid phase transformations on strength and vice-versa