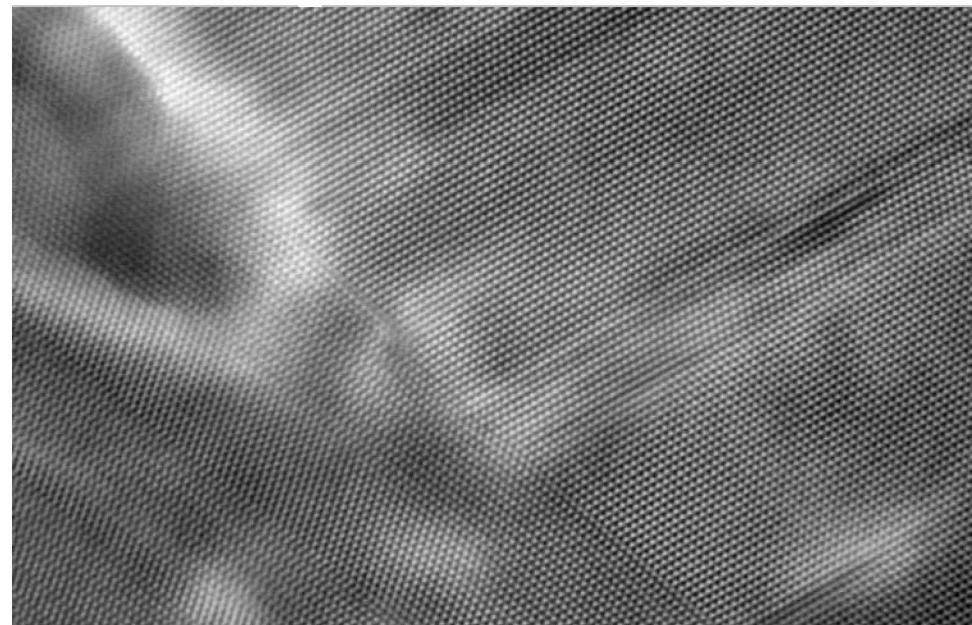


Electron Microscopy Research

Julian E.C. Sabisch

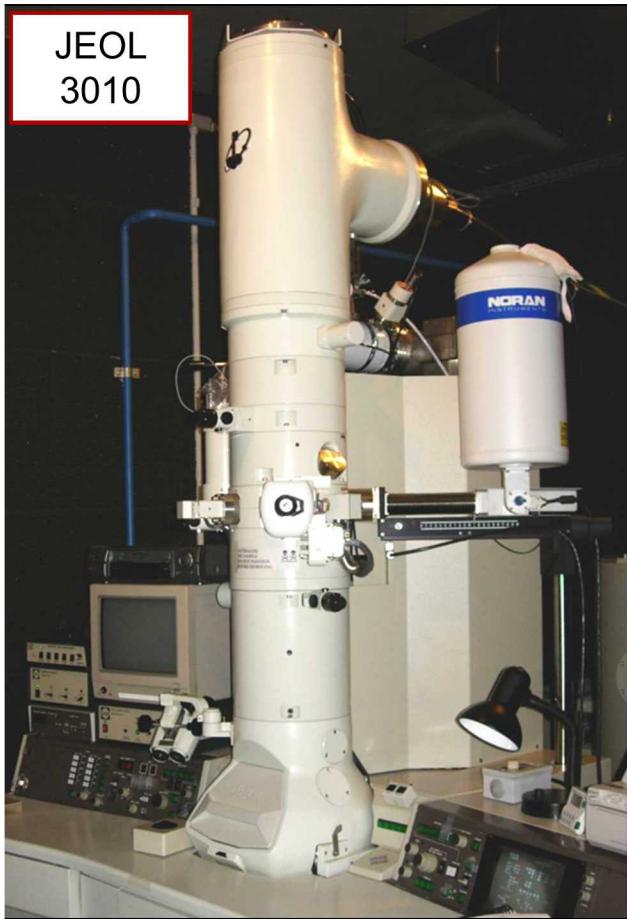
SAND2019-7706PE



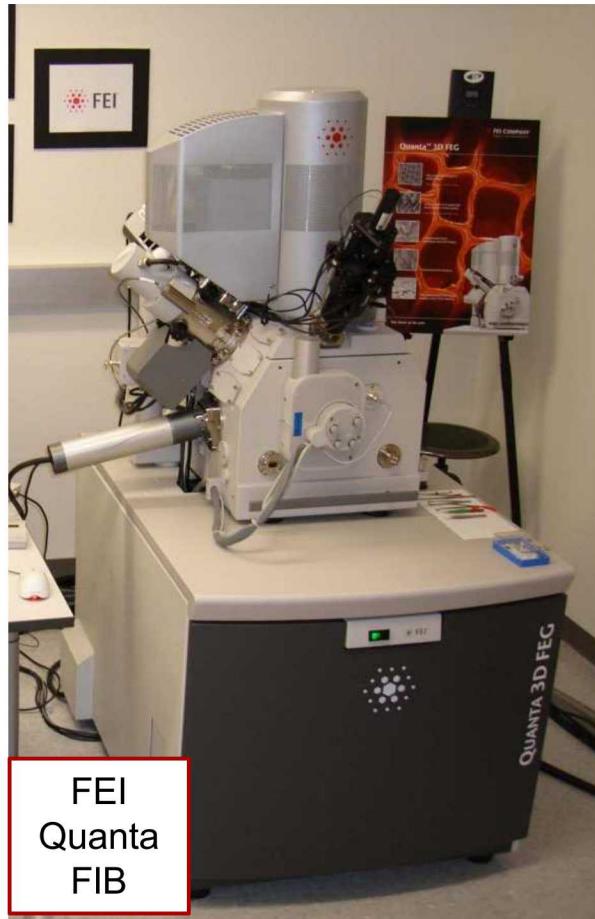
Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

Microscopy Overview

JEOL
3010



FEI
Quanta
FIB



Bruker D8
Powder
Diffractometer



Experienced on a wide range of microscopes including TEMs, SEM/FIBs and XRDs.

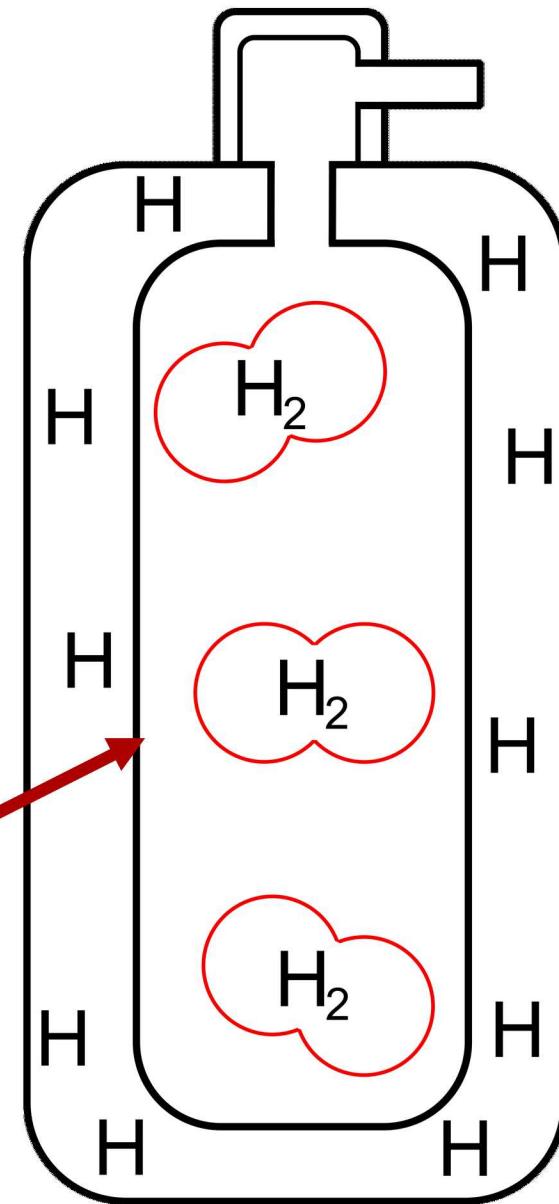
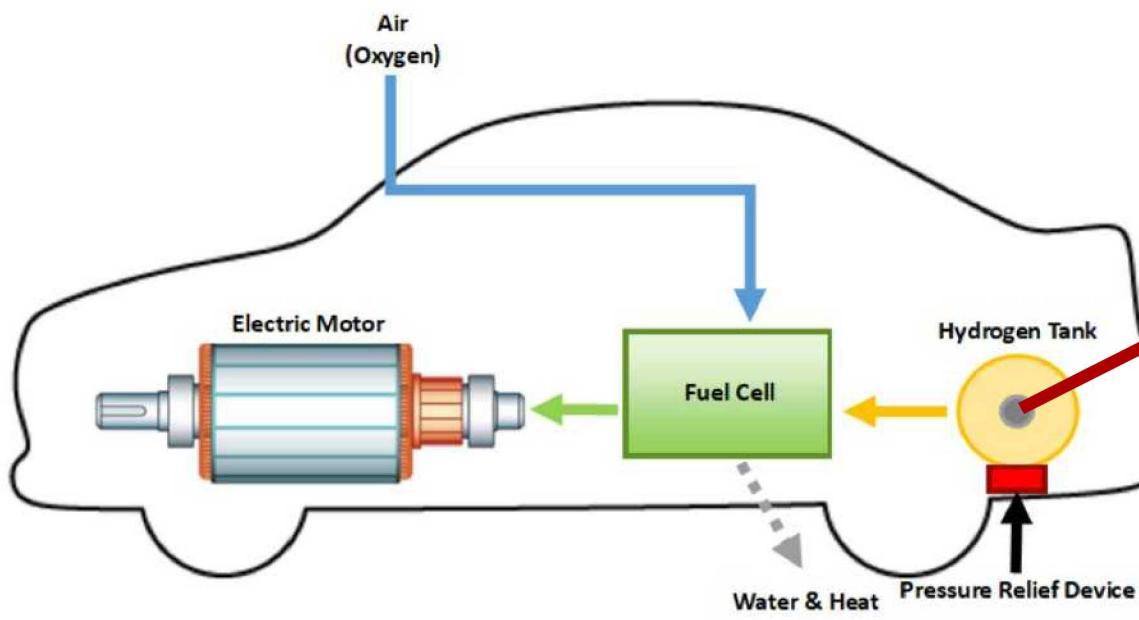
Microscope Overview

- Post-Doctoral research was performed on Sandia CA's new Thermo Fisher Themis Z probe corrected STEM/TEM.
- Multitude of capabilities, including:
 - Double tilt, Heating, Vacuum Transfer, Tomographic sample holders
 - Large voltage range from 60keV to 300keV for imaging a range of samples.
 - Direct electron detector (Gatan K2-SUI) integrated with energy filter.
 - High throughput large area x-ray detector for rapid EDX compositional mapping.

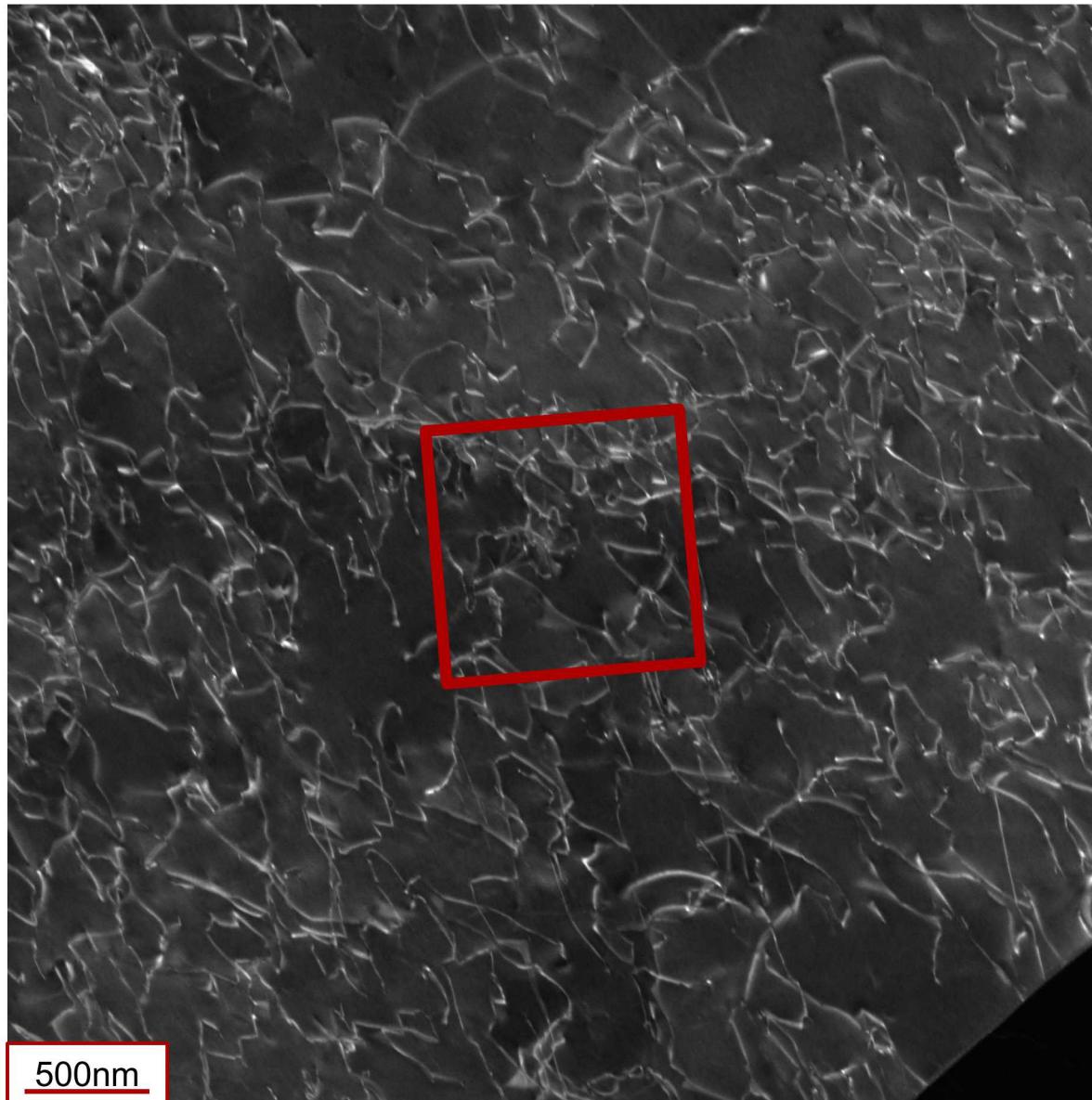


Interest in Hydrogen's Effect

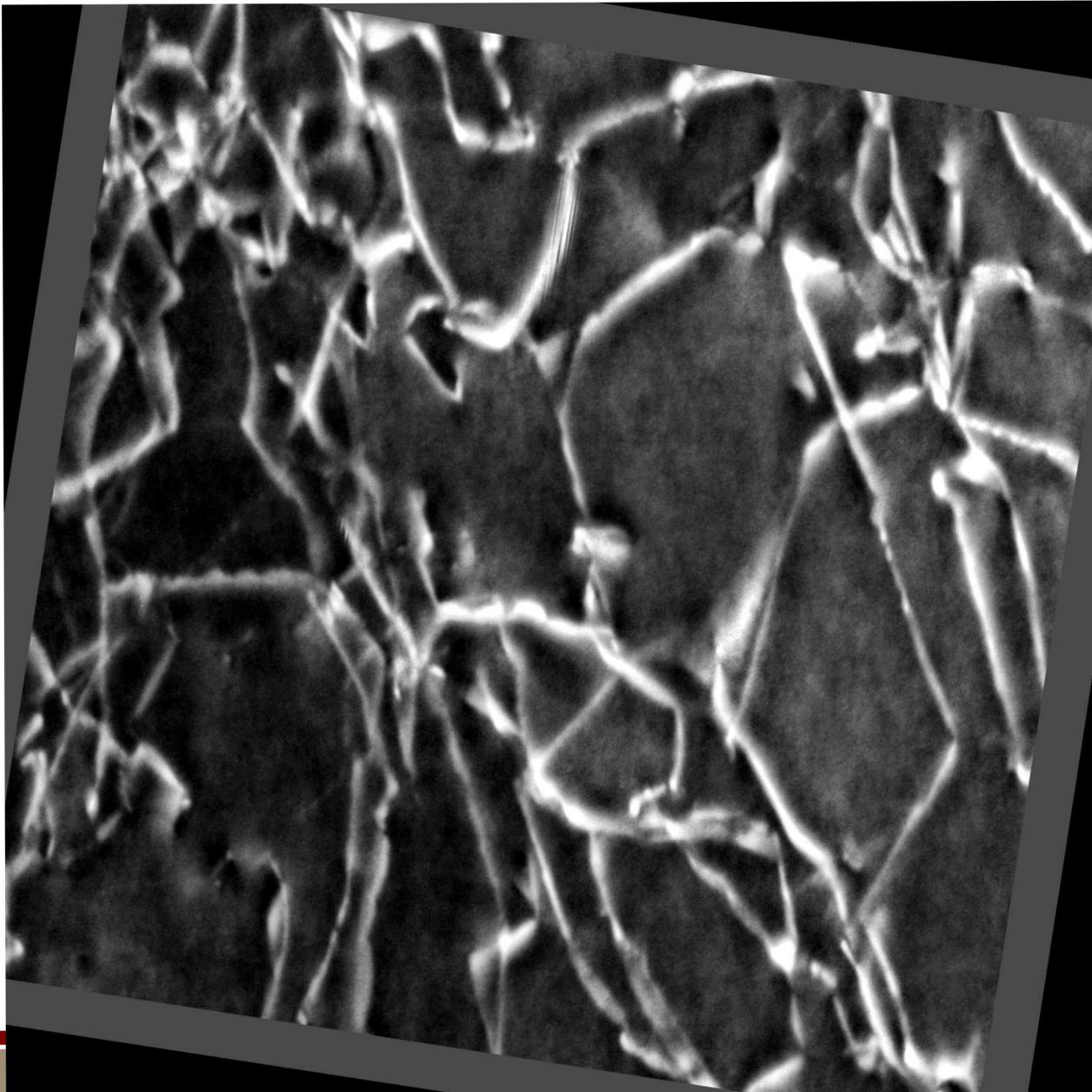
- Many metals have long been used to store hydrogen and its isotopes.
 - Fracture toughness has heavily decreased in hydrogen charged samples.
 - Understanding the effects of hydrogen on the microstructure and deformation behavior of metals has been a long term



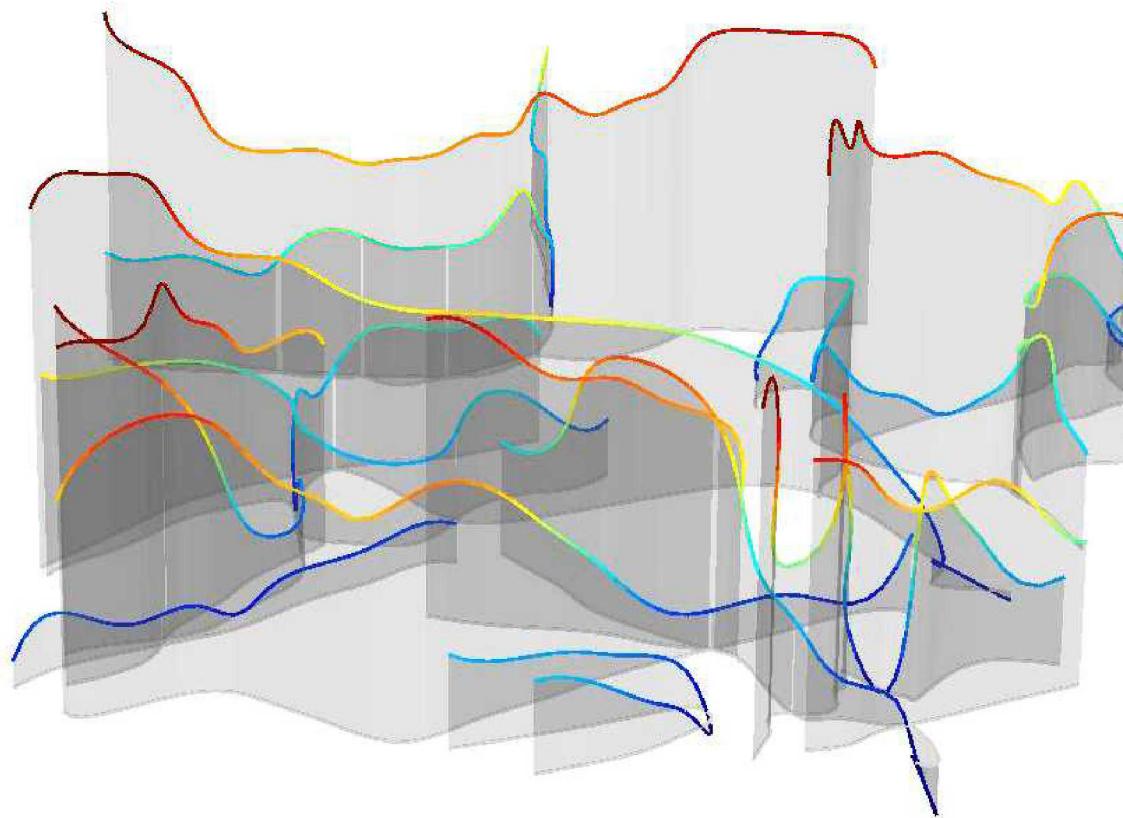
Collaboration on Tomography



Collaboration on Tomography

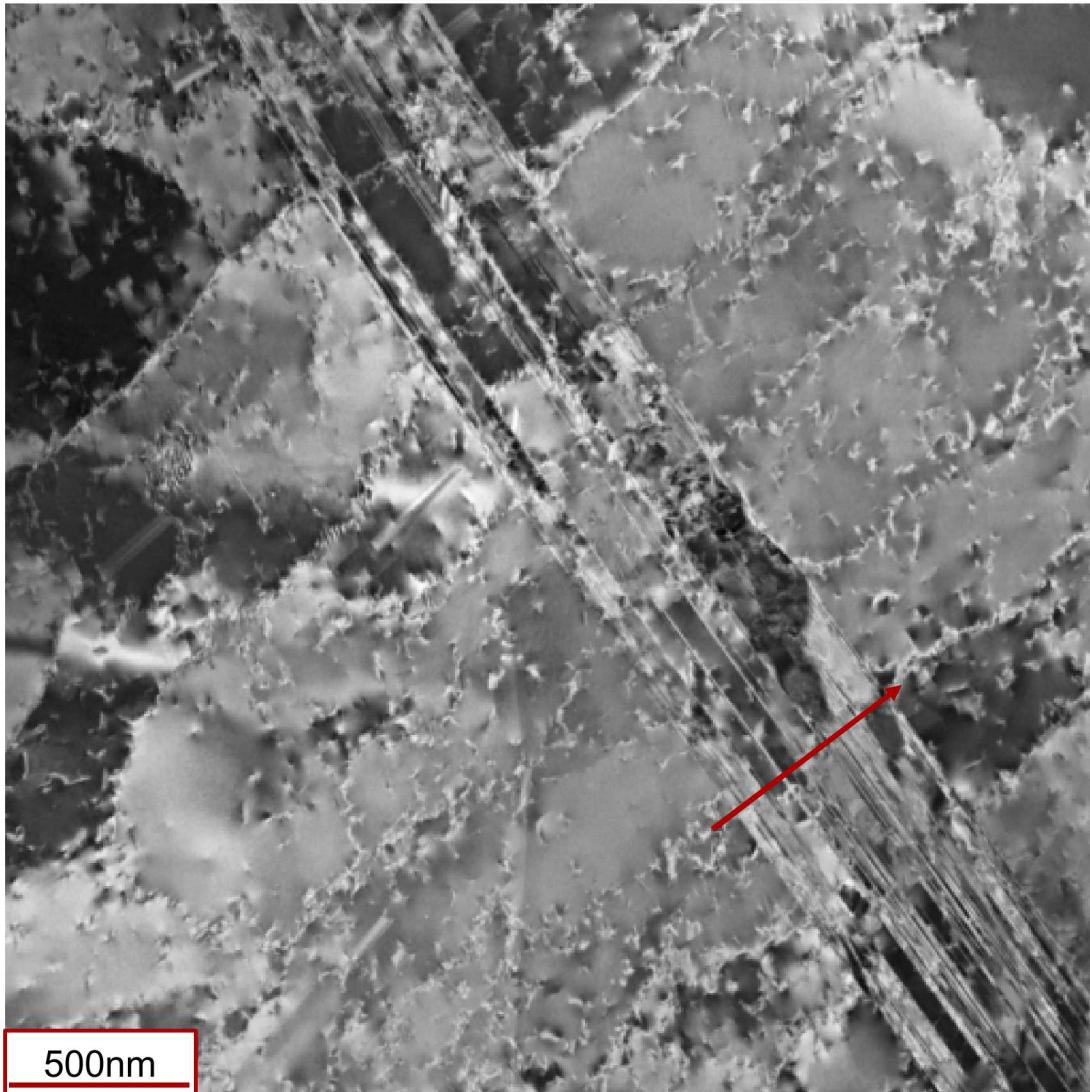


Collaboration on Tomography



Deformation Microstructure Mapping

DC-STEM Micrograph



500nm

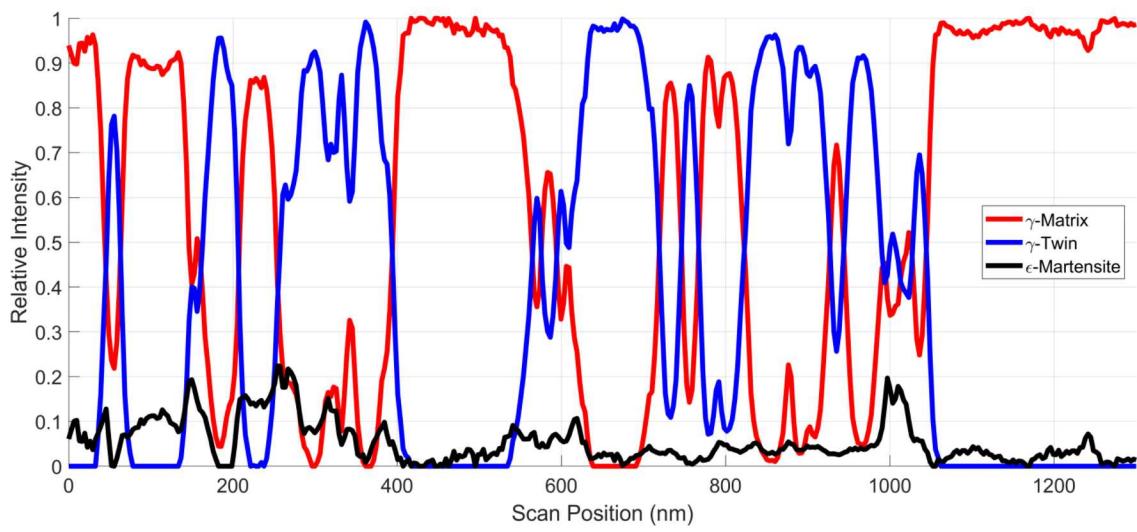
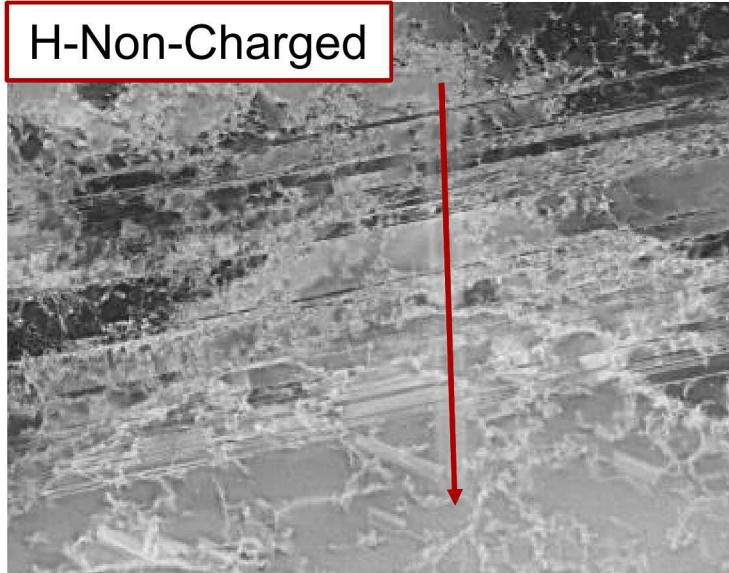
Single Line Scan ~200 Patterns



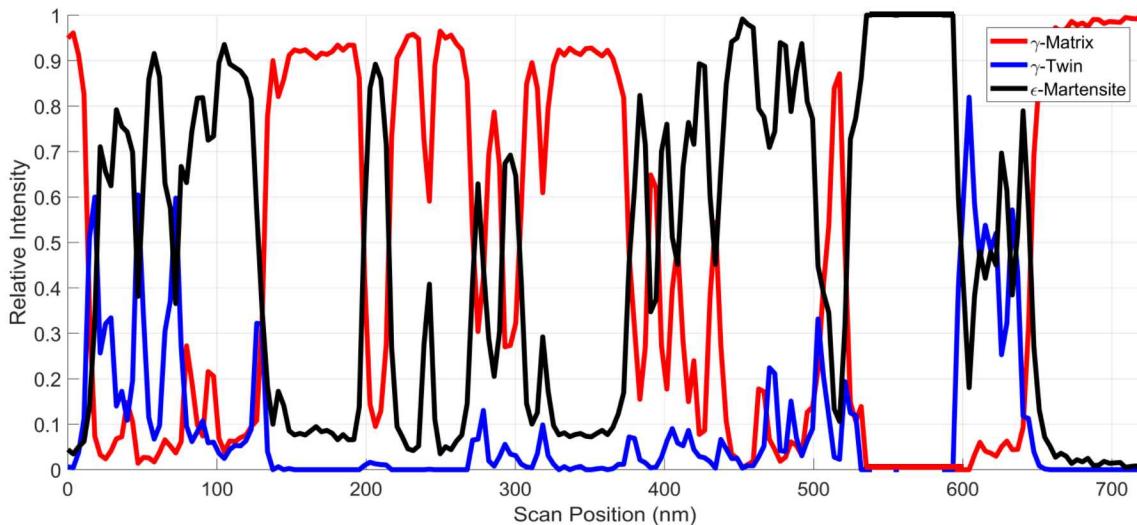
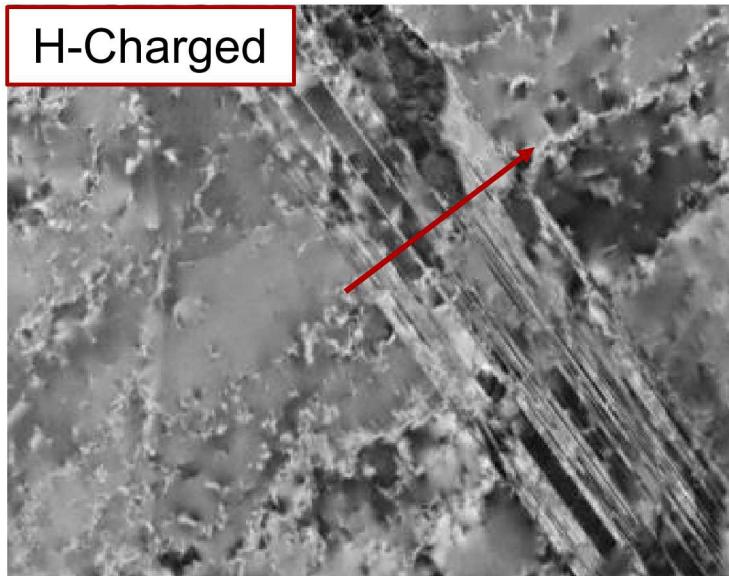
In H-Charged samples ε -martensite dominates within the deformed region, with little dislocation content observed.

Quantitative Phase Measurements

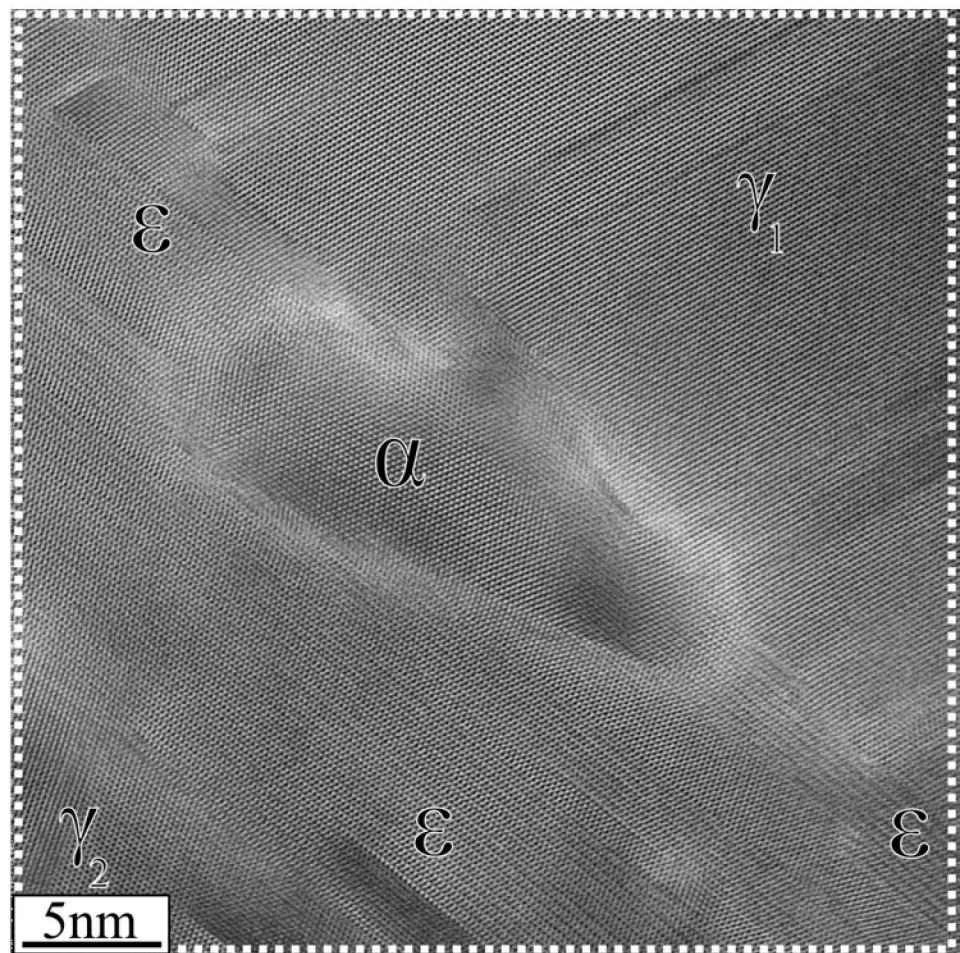
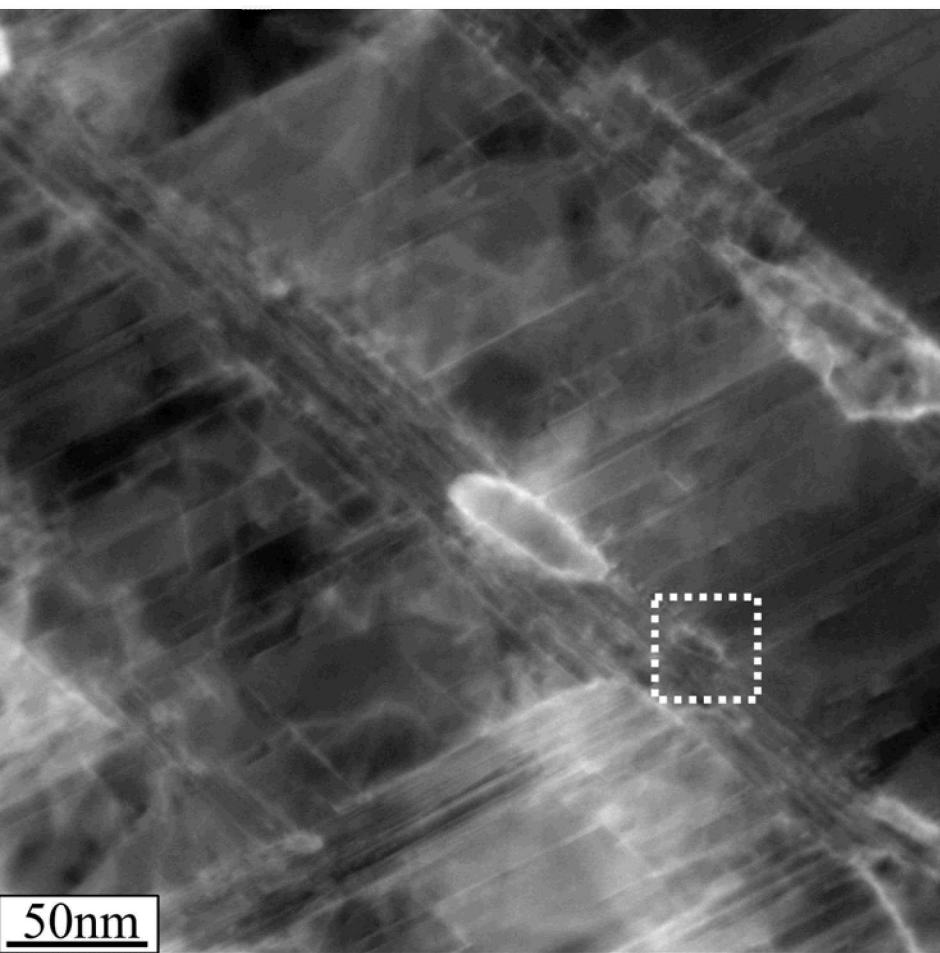
H-Non-Charged



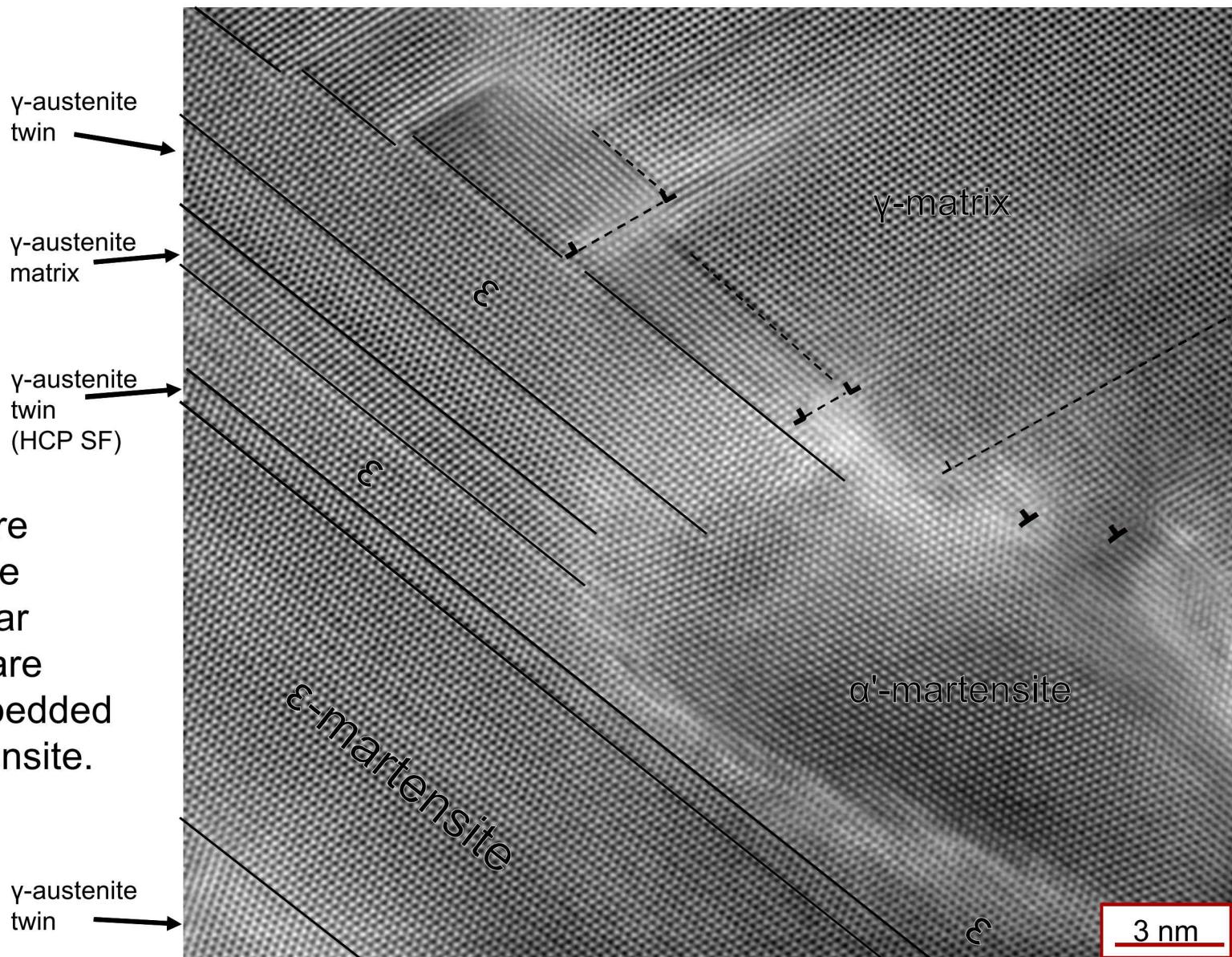
H-Charged



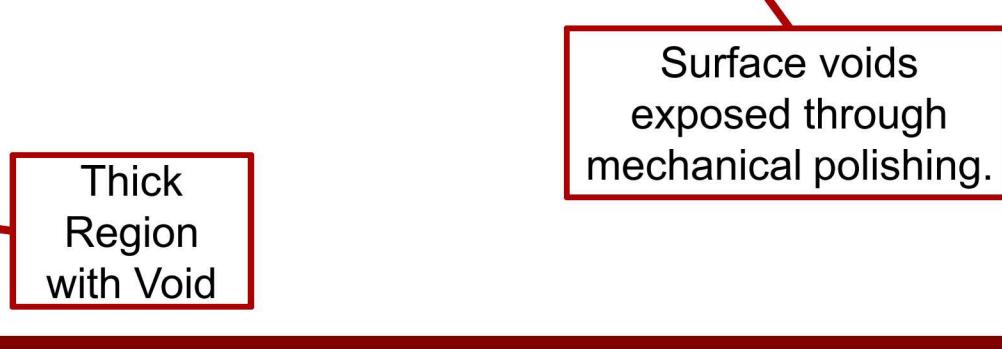
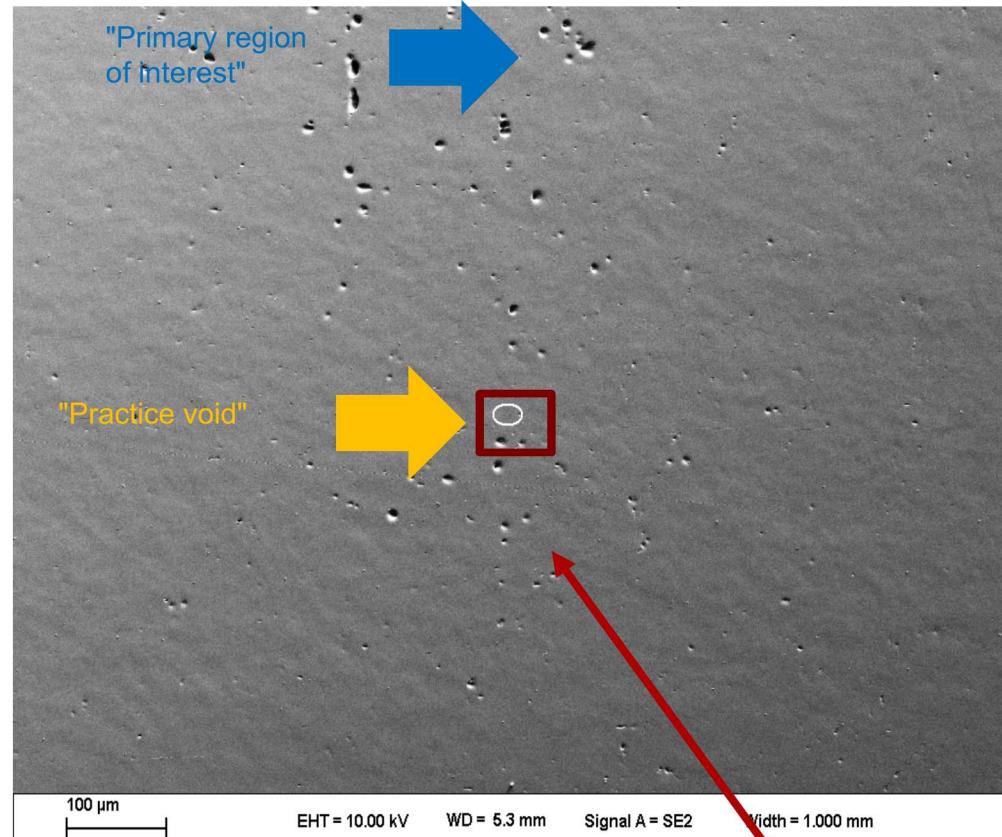
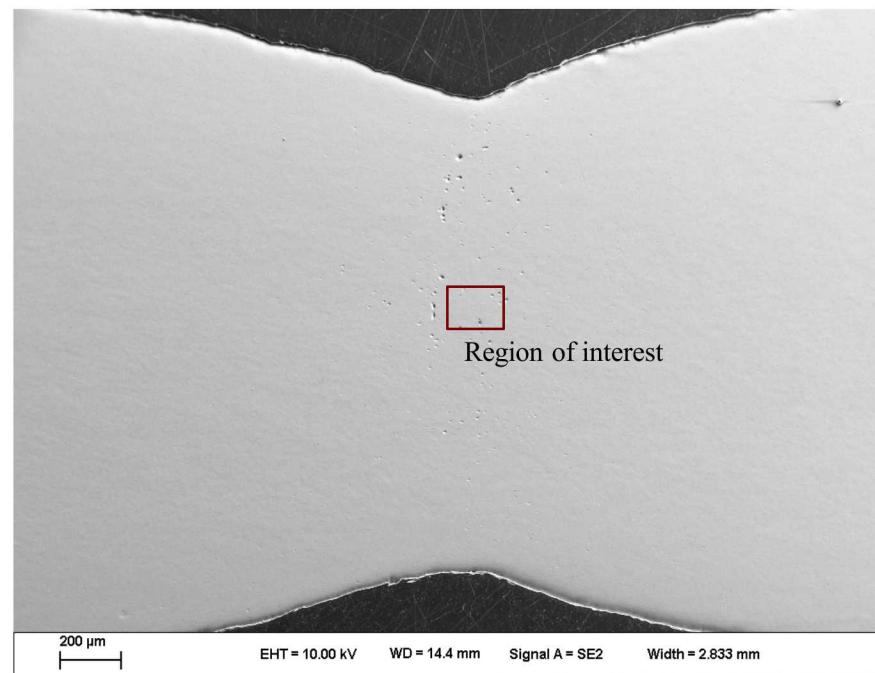
Shear Bands in H-Charged 20% Strain



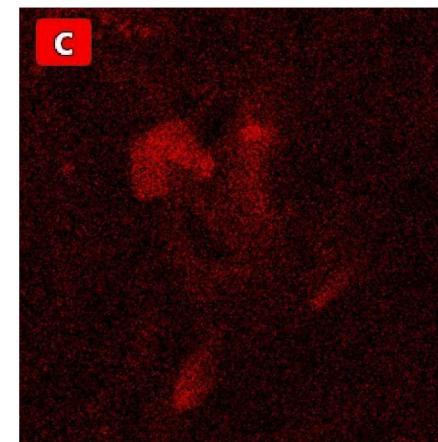
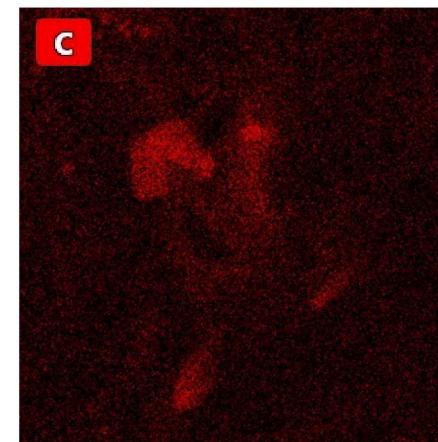
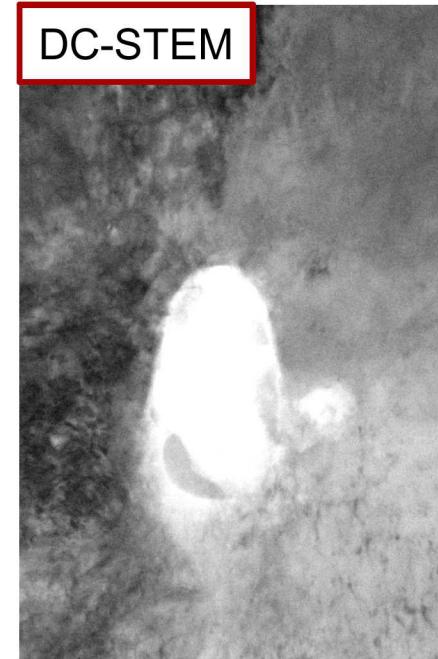
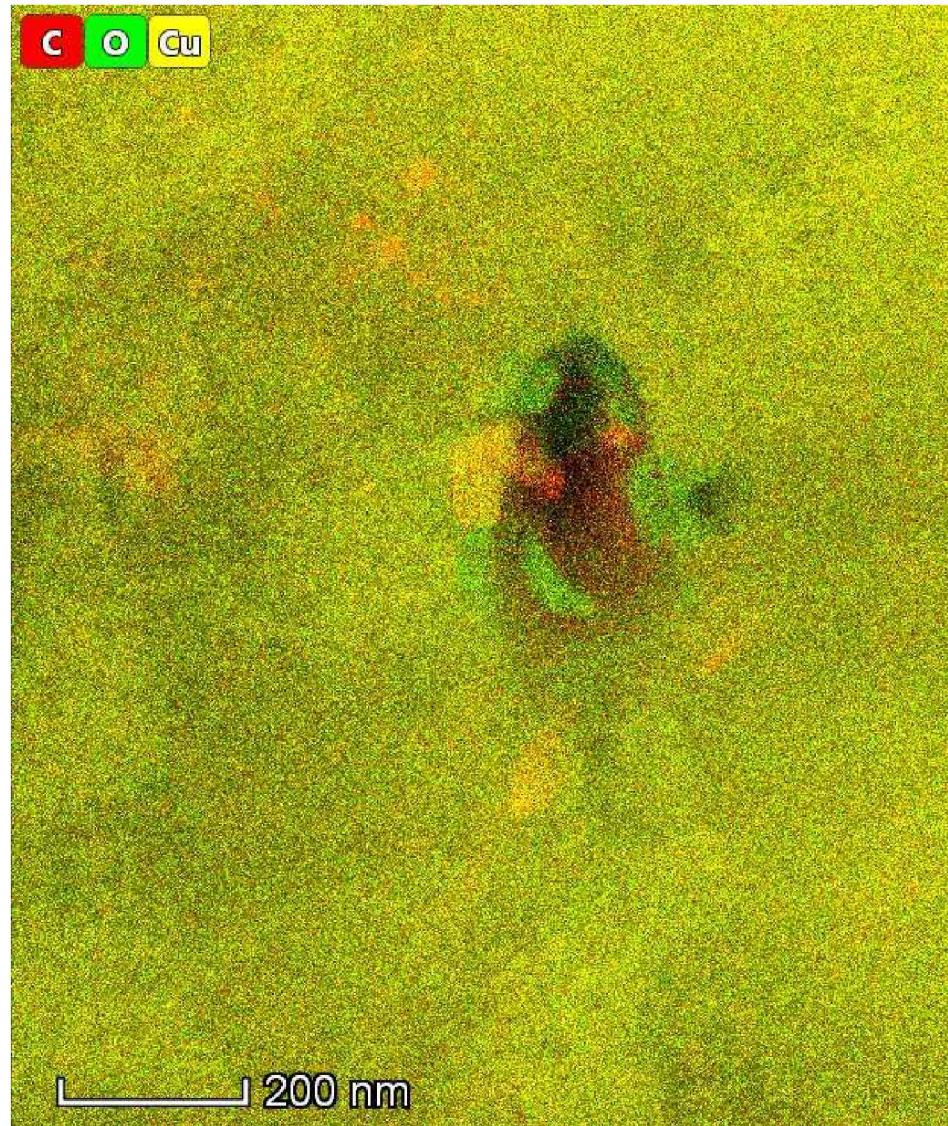
20% HC HRSTEM



Ductile Rupture of Copper

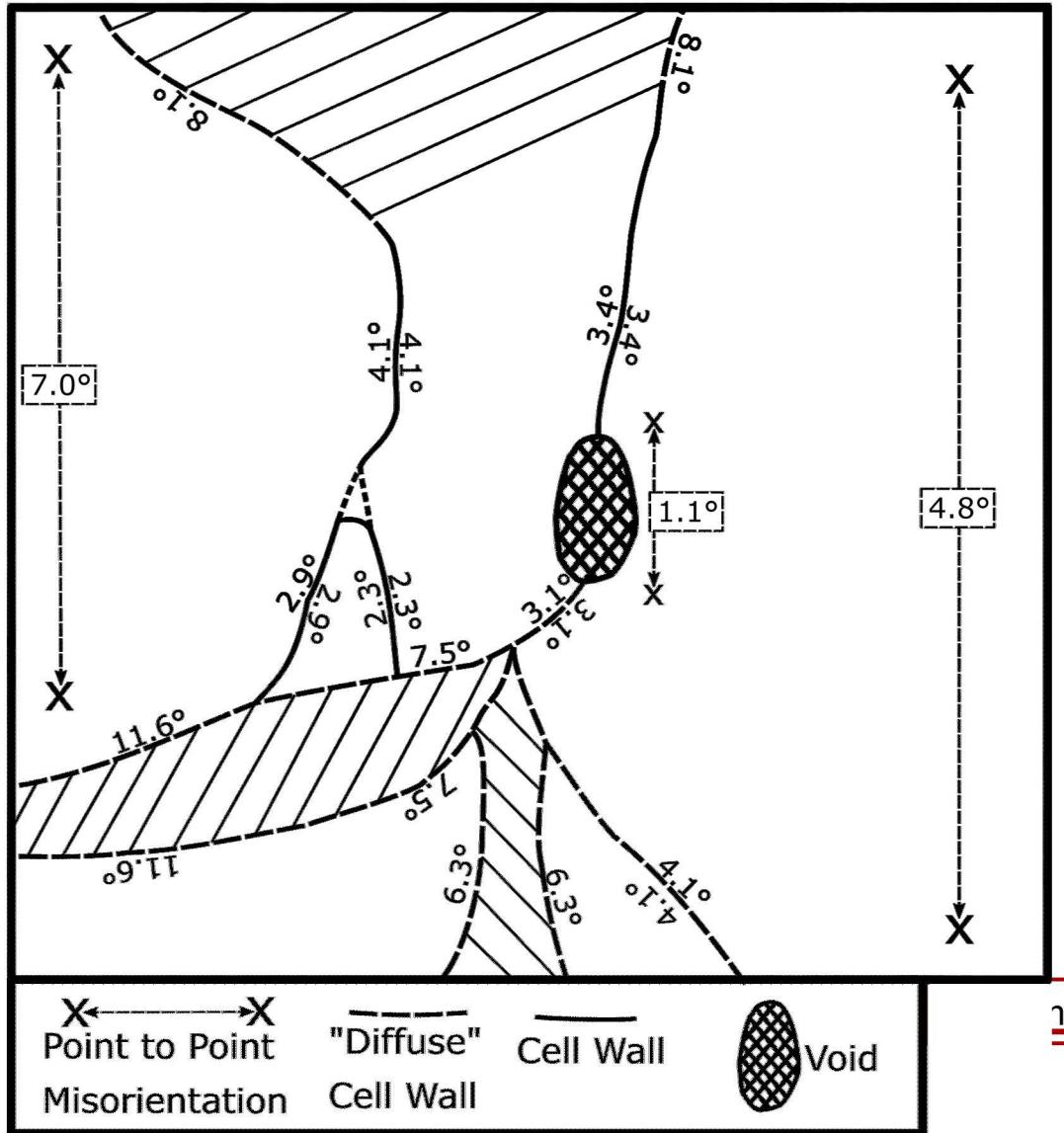


EDX of Void



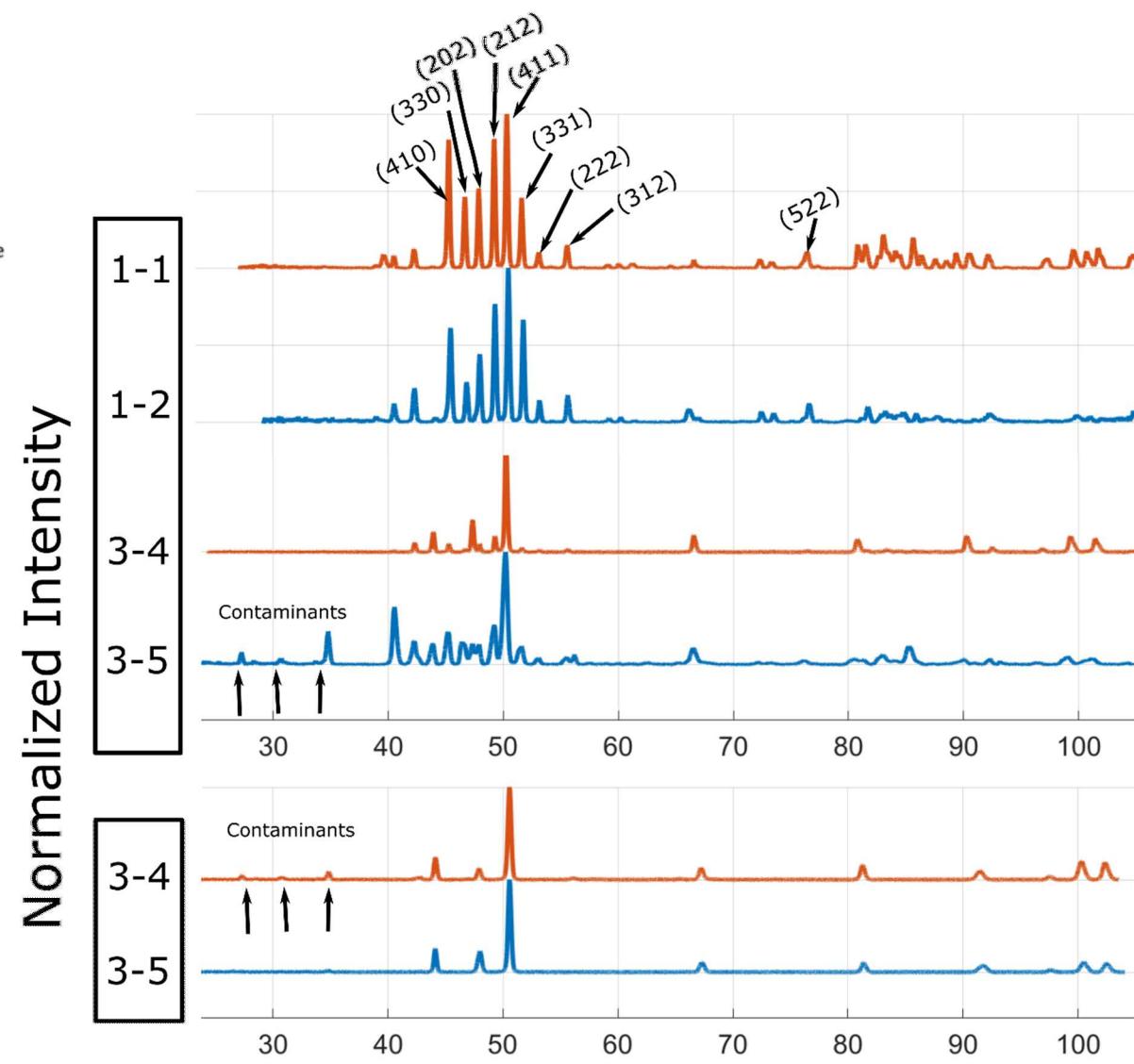
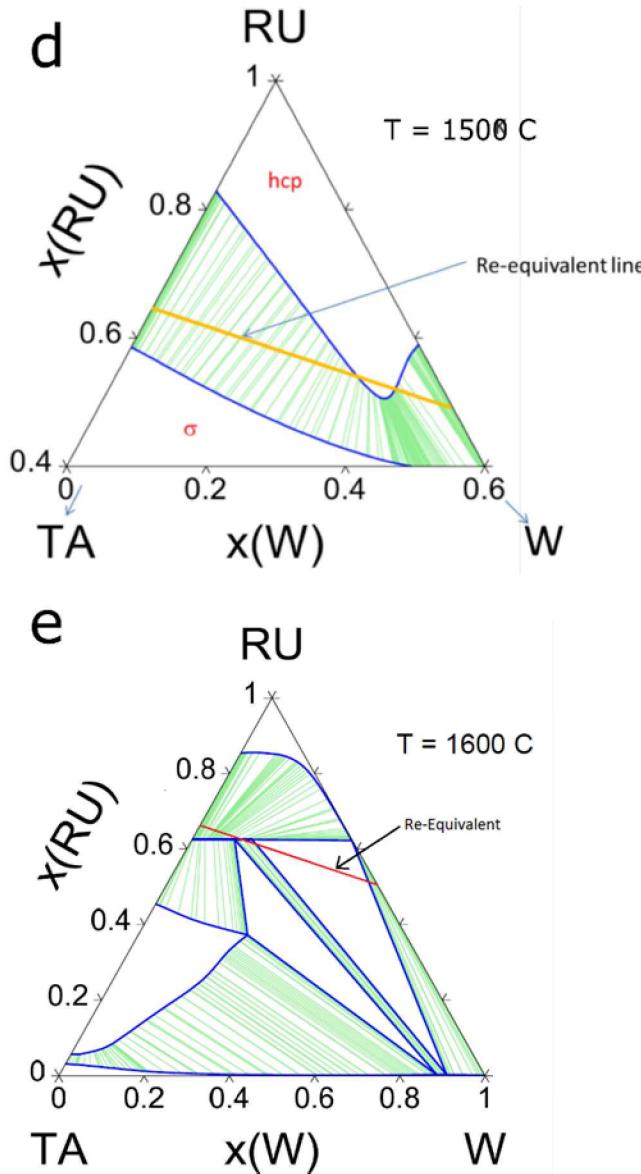
- High concentrations of Oxygen and Carbon inside the void.
- SAD patterns show some contraction of diffraction spots, indicating the presence of Cu_2O .
- FIB preparation was determined to not be the cause of oxygen content within the void.

Schematic of Orientation



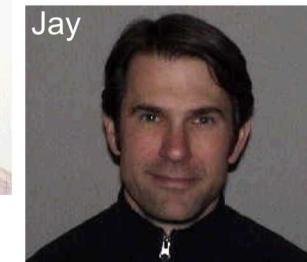
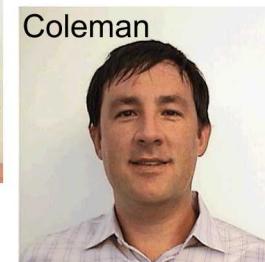
- Select diffraction patterns were chosen in order to manually identify the misorientation change surrounding the void.
- Cell block boundaries appear likely to become “diffuse” (not seen a few microns from the void).
- Diffuse boundaries show large misorientation changes ($>6^\circ$) over the span of a few hundred nanometers.
- Misorientations appear larger further from the void.
- Void walls are orthogonal to the $\{224\}$ and $\{111\}$ reflections.

XRD Analysis in Alloy Development

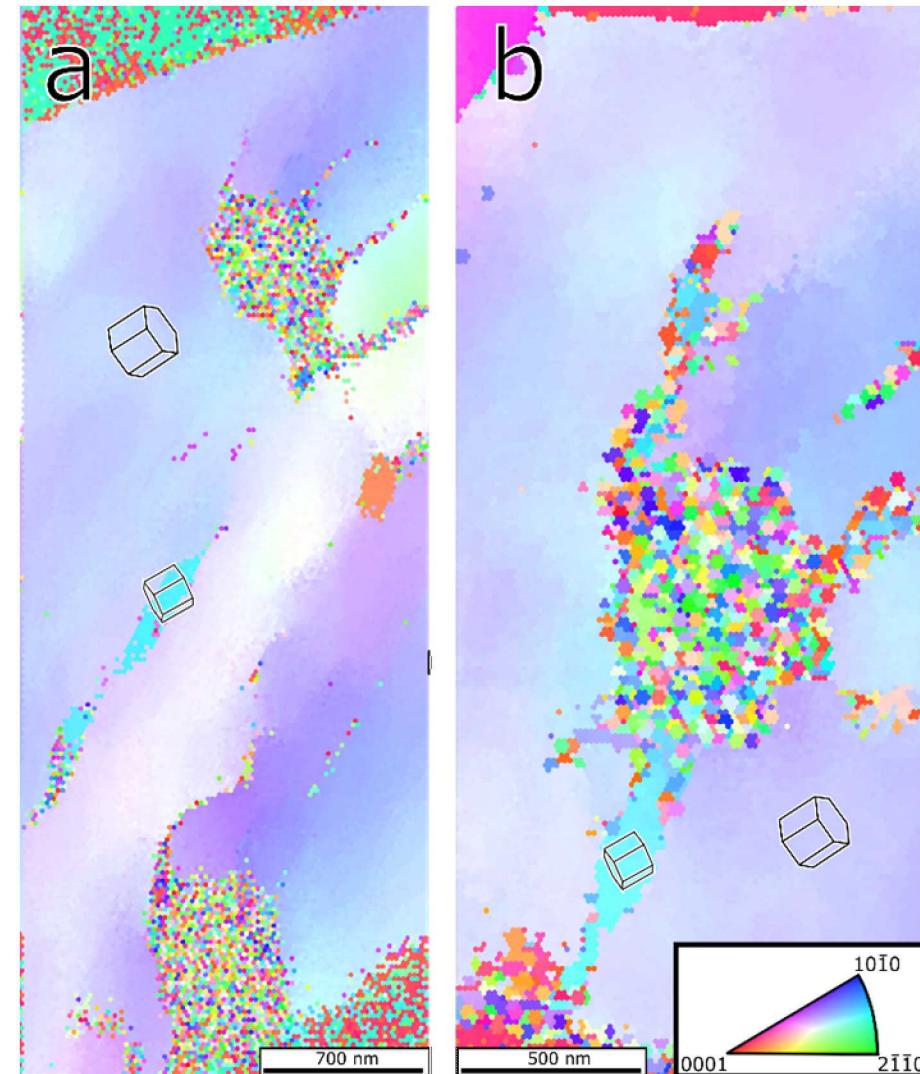
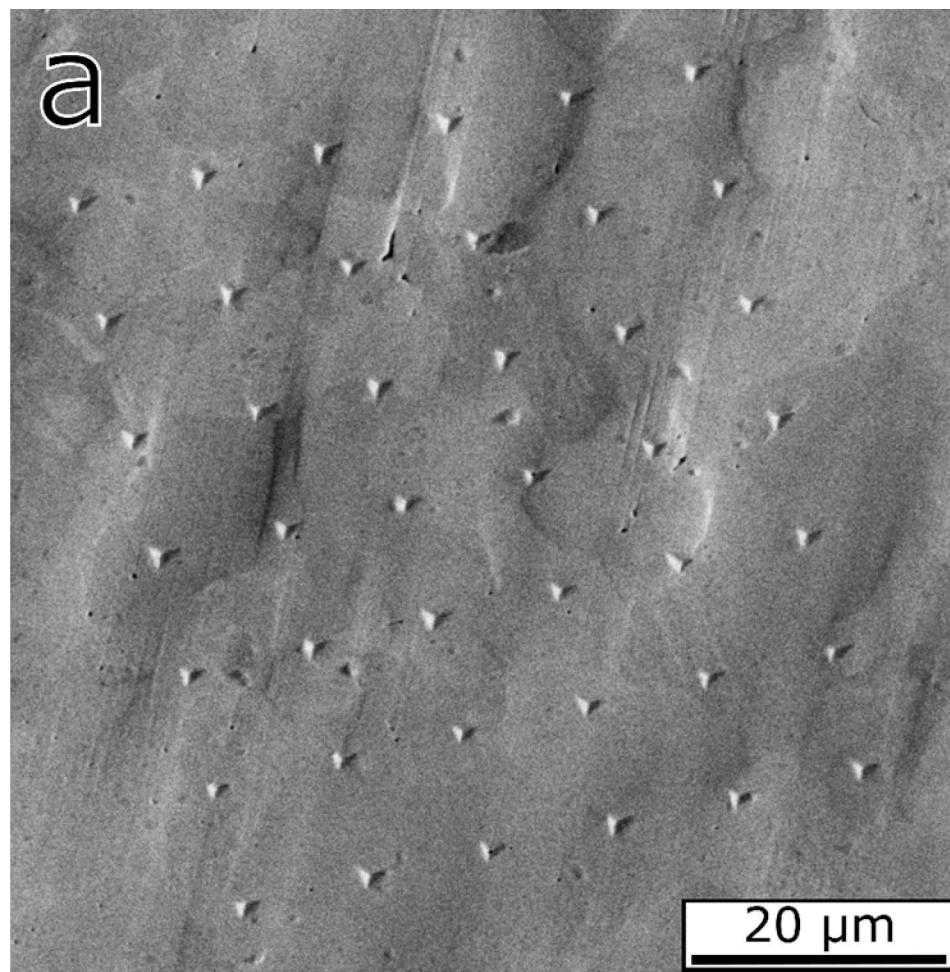


Thank you

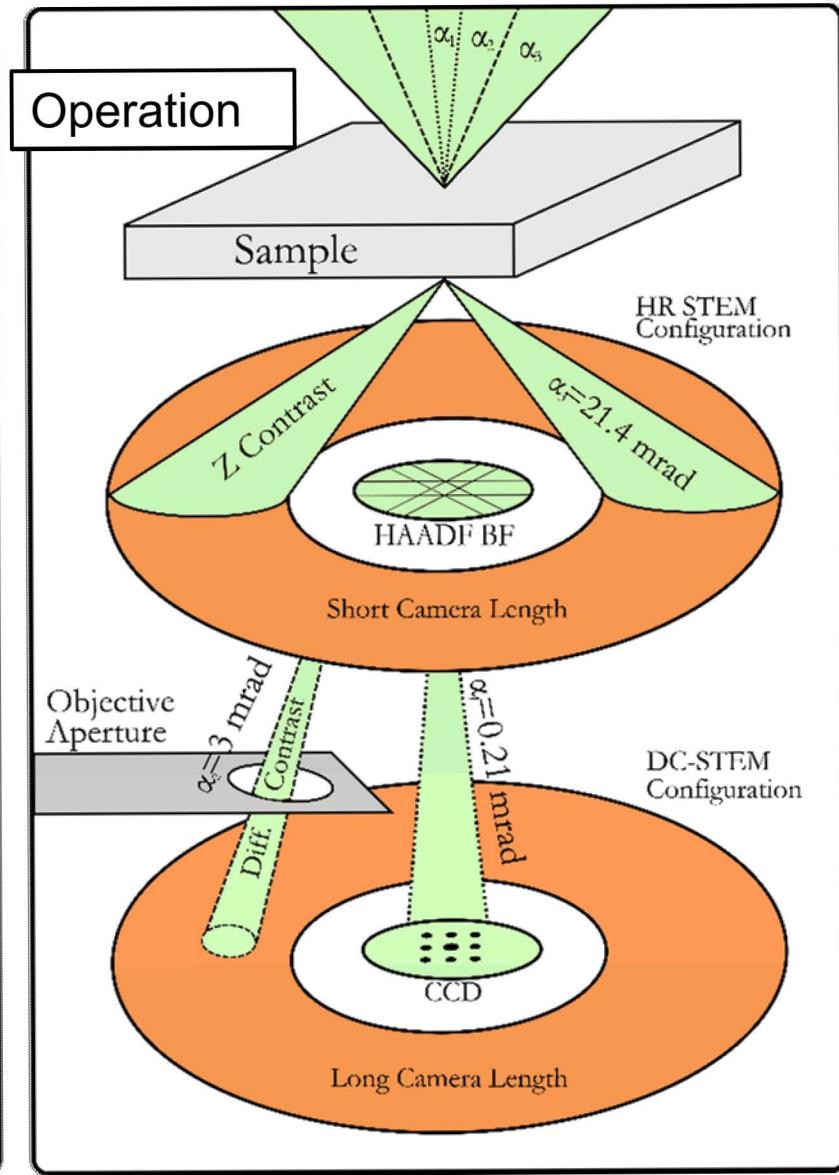
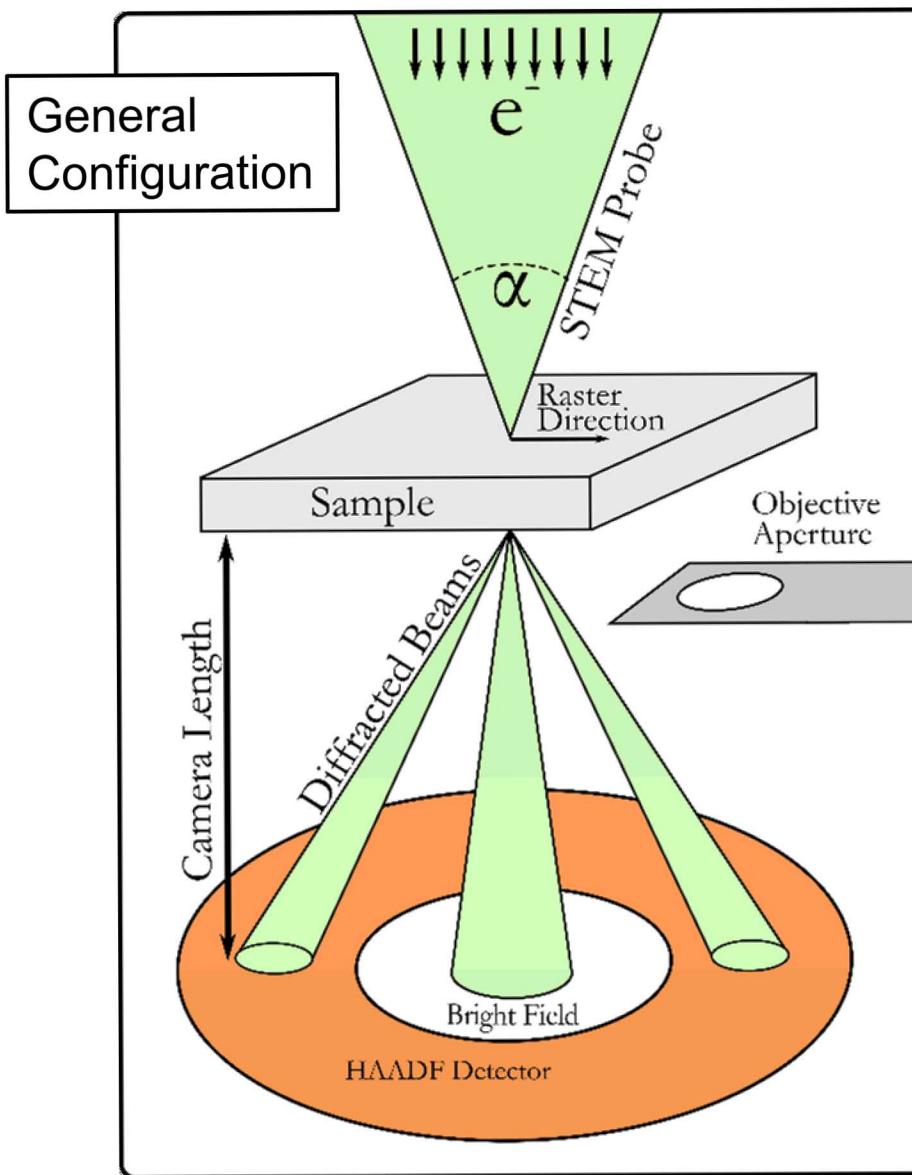
- Thank you for your attention!
- Thanks to: Mark Homer, Warren York, Heidy Vega.
- Microscopy and Microanalysis:
 - Doug Medlin
 - Josh Sugar
- Mechanical Testing and Materials
 - Chris San Marchi
 - Joe Ronevich
 - Dorian Balch
- Modeling of Materials
 - Ryan Sills
 - Coleman Alleman
 - Jay Foulk



XRD Analysis used for EBSD

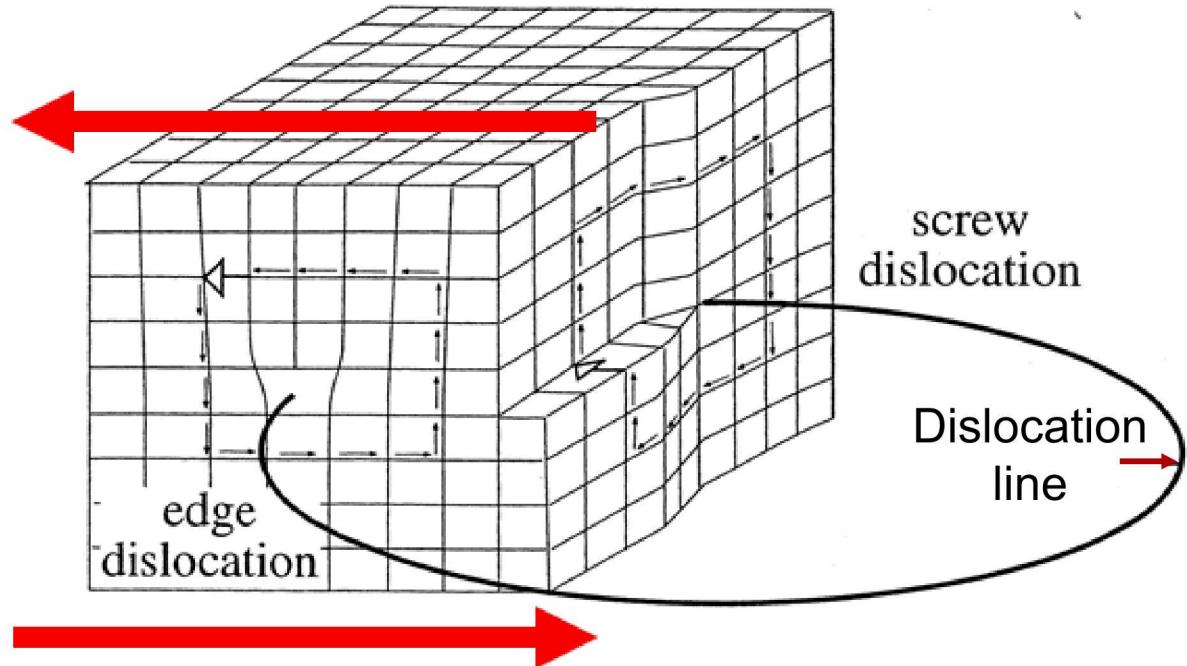


Background on STEM



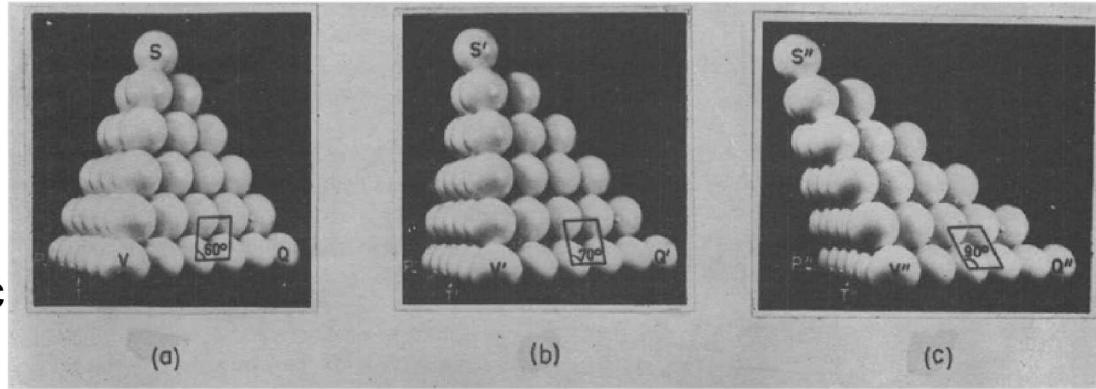
Background on Dislocations

- Plastic deformation in metals is largely accommodated through dislocations.
- Deformation characteristics can change based on dislocation character.
- Dislocations can have edge, screw, or mixed type.



Background on Twinning

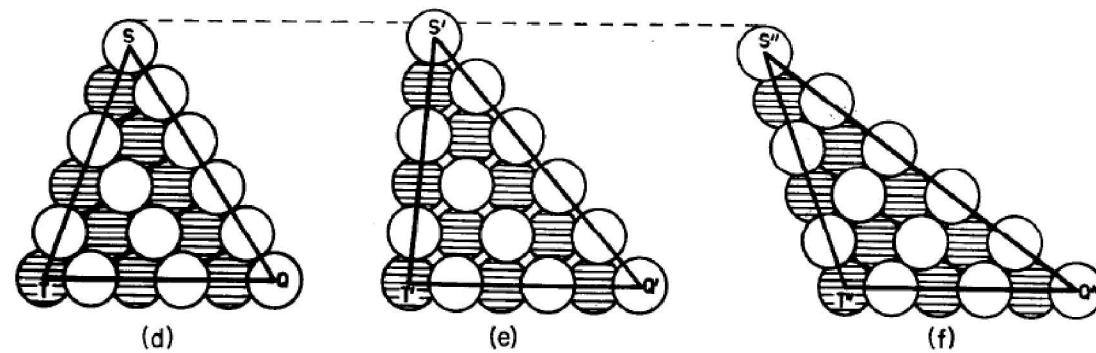
- Metals can also deform through twinning.
- Twinning is when one part of a crystal is oriented with respect to another through a crystallographic symmetry operation.
- In the case of Austenitic Steels twinning is induced through Deformation and produces a Mirror symmetry along $\{111\}$ planes.
- A $\langle 112 \rangle$ shear along every $\{111\}$ plane produces a twin, the shear spread to every other $\{111\}$ plane produces an HCP structure



(a) Original tetrahedral arrangement with four $\{111\}$ planes.

(b) Intermediate position after $1/3$ of normal twin shear. The distance between successive $\{111\}$ planes parallel to $P'V'S'$ and $P'V'Q'$ has increased by 5.4% .

(c) Final position after complete twin shear.

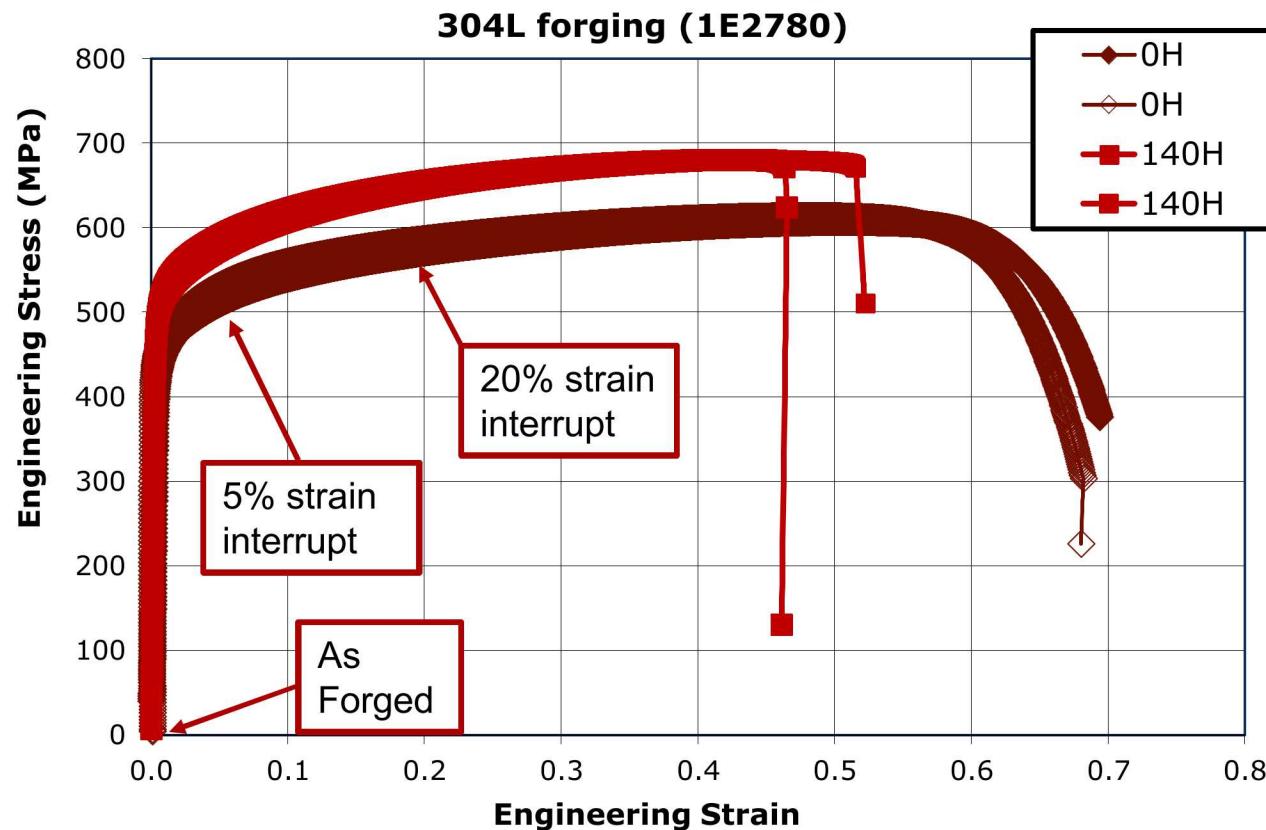


(d-e-f) Cross section through the $\{110\}$ plane TQS perpendicular to PV.

FIG. 2. Normal twin shear in a cubic close-packed arrangement of spheres. The horizontal $\{111\}$ planes are sheared in the $\langle 112 \rangle$ direction perpendicular to PV.

Composition and Mechanical Data

MCN	Fe	Cr	Ni	Mn	Si	C	N	P	S
200956	Bal	19.64	10.6	1.62	0.65	0.028	0.04	0.02	0.0042
Test Temp		Nominal Hydrogen concentration		Yield strength (MPa)		Tensile strength (MPa)		Uniform Elongation (%)	
RT		0 140		436 488		611 680		63.2 70.8	



Dislocation Tomography

FIB lift-outs of pure copper sample deformed to necking.

Here we see the cell block boundaries (formed through dislocation networks)

