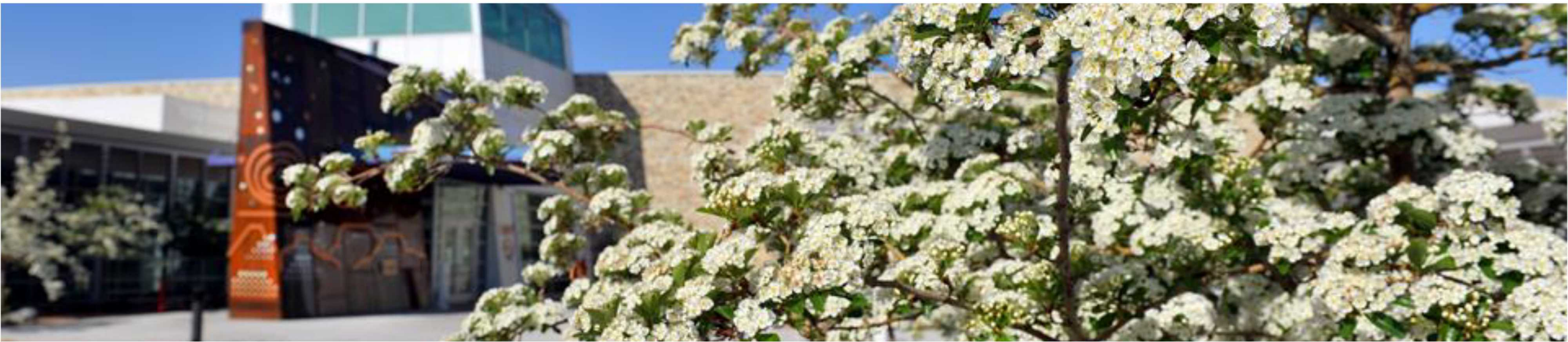


Large-Scale Molecular Dynamics (MD) Derivation of Length-Insensitive Dislocation Mobility Laws in Fe-Ni-Cr Stainless Steels

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Motivation

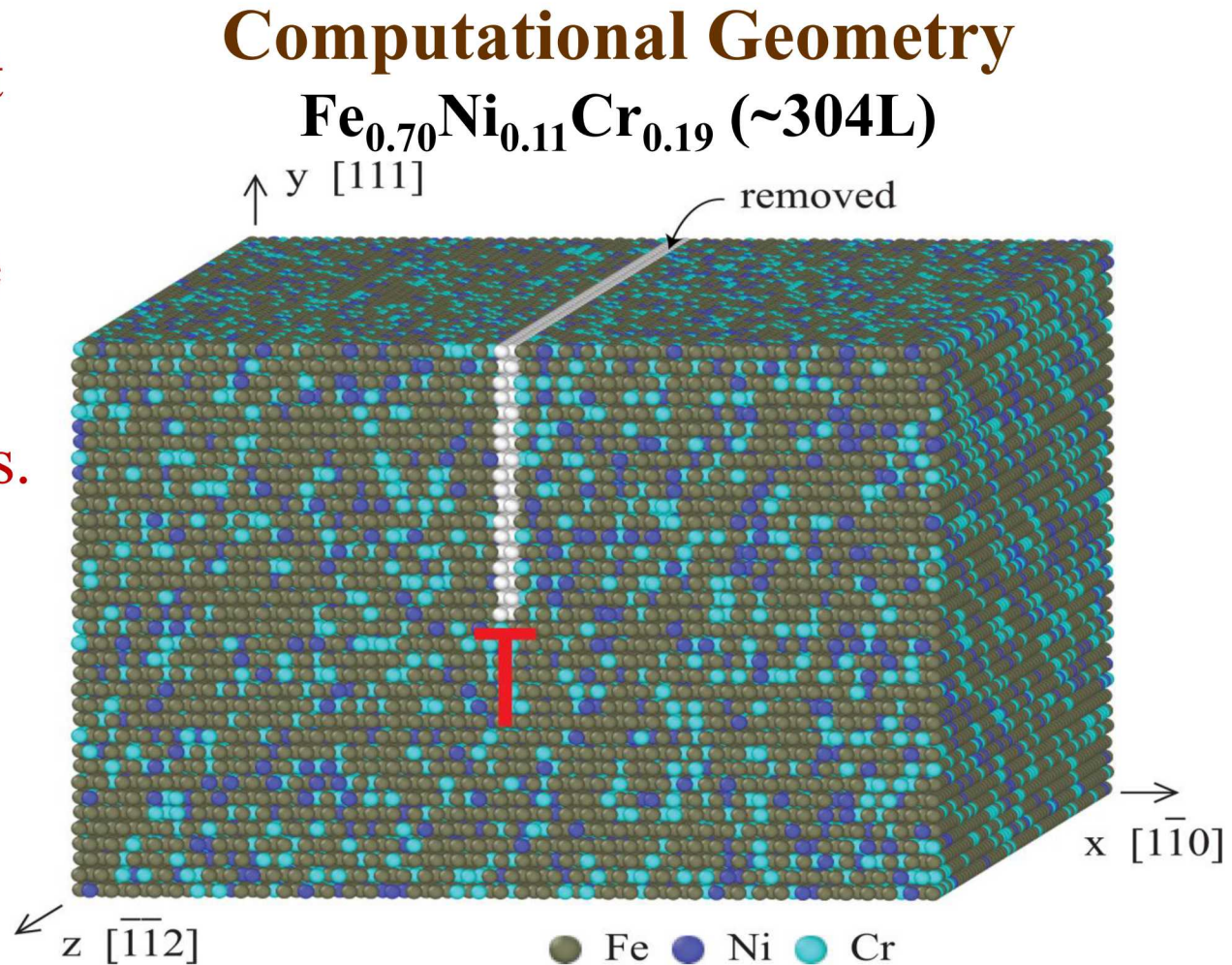
- ❑ Dislocation mobility laws are required for dislocation dynamics simulations but cannot be measured experimentally.
- ❑ Dislocation mobility in alloys is indicative of solute strengthening.
- ❑ Stainless steels are important structural materials.
- ❑ Past MD results of dislocation mobility in alloys are size sensitive.
- ❑ Mobility laws in a wide range of stresses and temperatures have not been well understood.
- ❑ **We attempt to understand size-insensitive mobility laws in a wide stress-temperature space for stainless steels.**

Goals

- ❑ Use MD to systematically study edge dislocation mobility in random $\text{Fe}_{0.7}\text{Ni}_{0.11}\text{Cr}_{0.19}$ over a range of stresses and temperatures.
- ❑ Develop a theoretical mobility model to rationalize the MD data.
- ❑ Understand when and why the mobility becomes length-dependent.

Approach

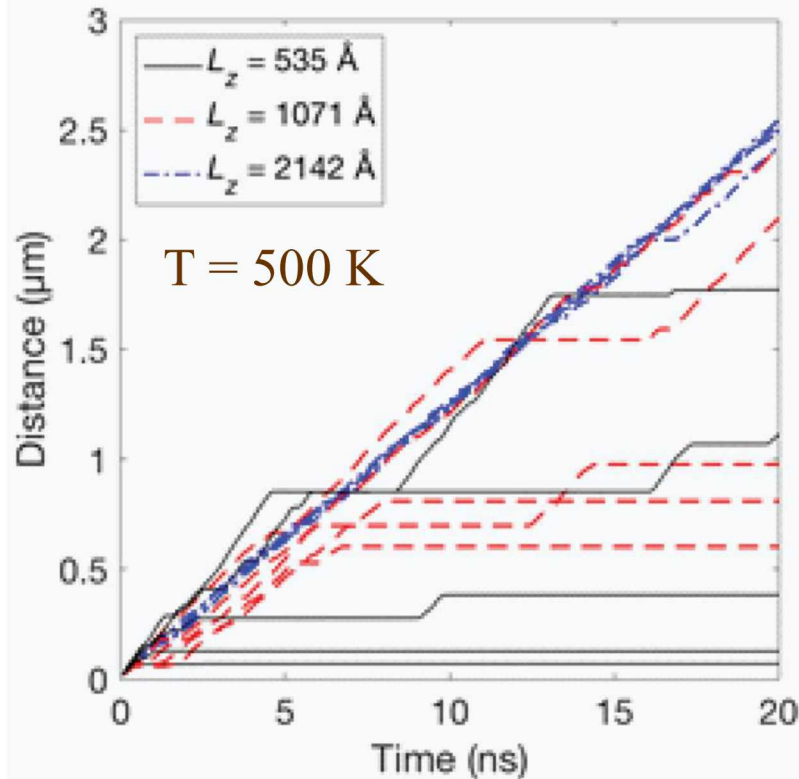
- ❑ Use an Fe-Ni-Cr EAM potential (Zhou et al, 2018).
- ❑ Stresses are modeled as forces on surface layers of atoms (in $\pm x$).
- ❑ NPT simulations are performed for ≥ 8 ns.
- ❑ Dislocation distance is determined by scaling the relative shift between upper / lower halves of the crystal.
- ❑ For each stress / temperature condition, various sizes are explored.



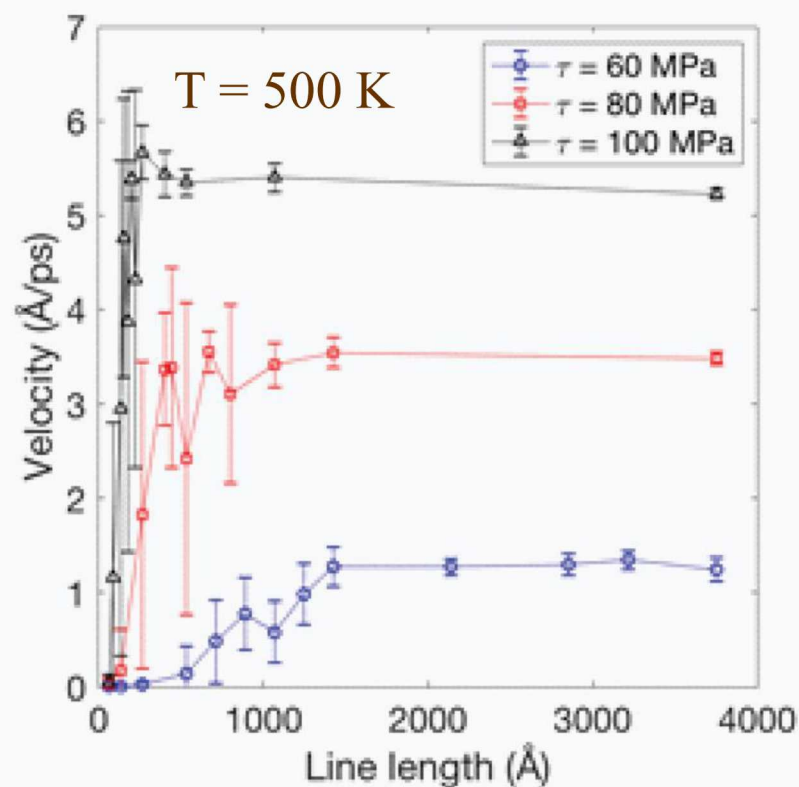
L_z : dislocation length, L_y : dislocation-surface spacing, L_x : periodic moving distance.

Size Effects

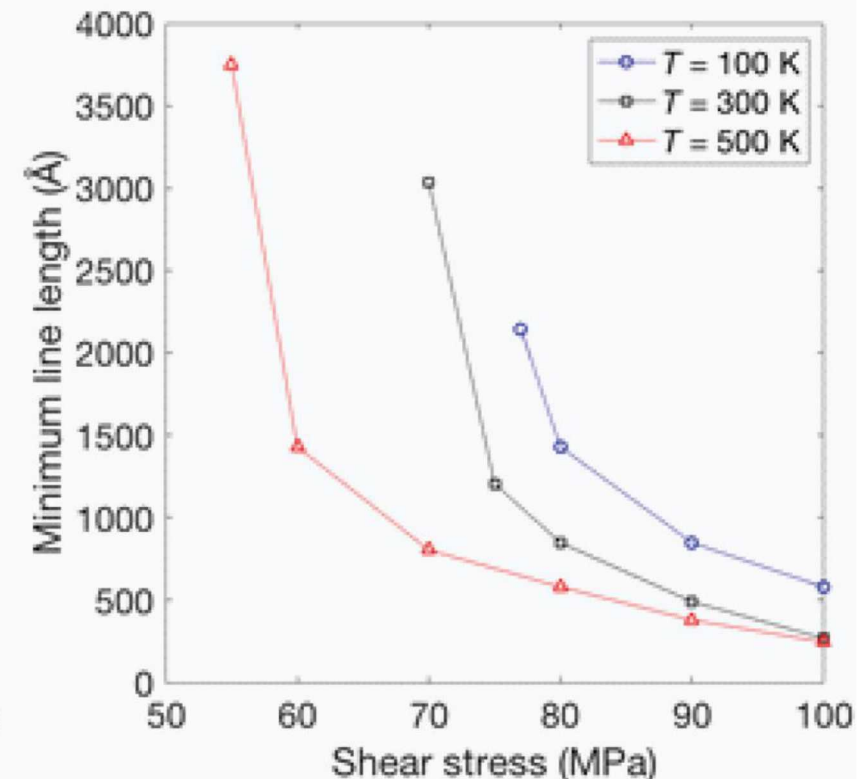
(a) distance at $\tau = 60$ MPa



(b) velocity



(c) threshold size-independent length

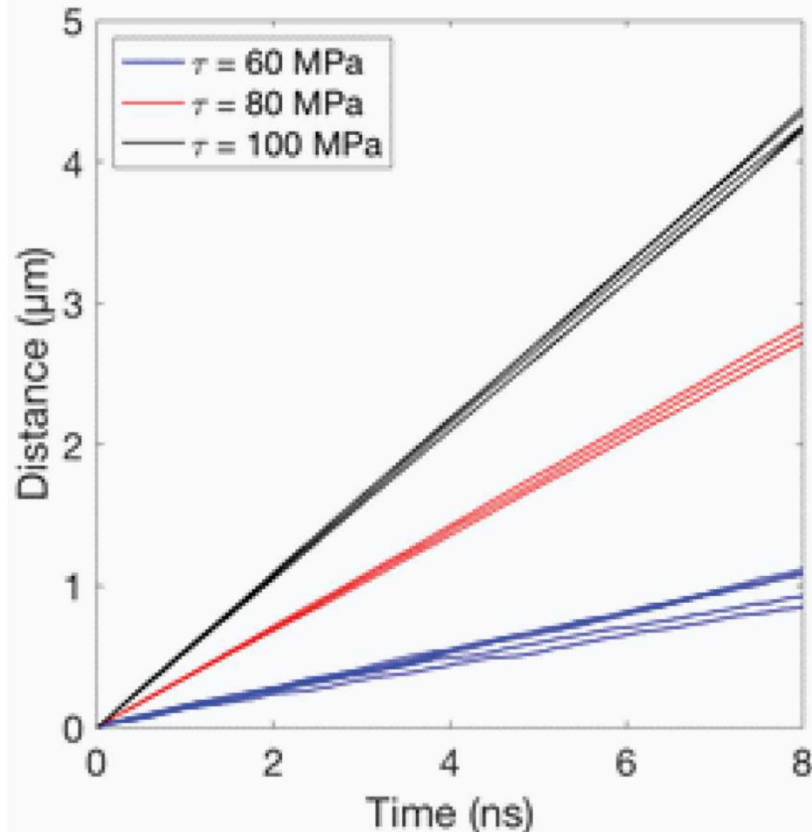


- ❑ Velocities are sensitive to dislocation length L_z but not to L_y and L_x .
- ❑ Size-insensitive velocities can be obtained at high stresses and temperatures.

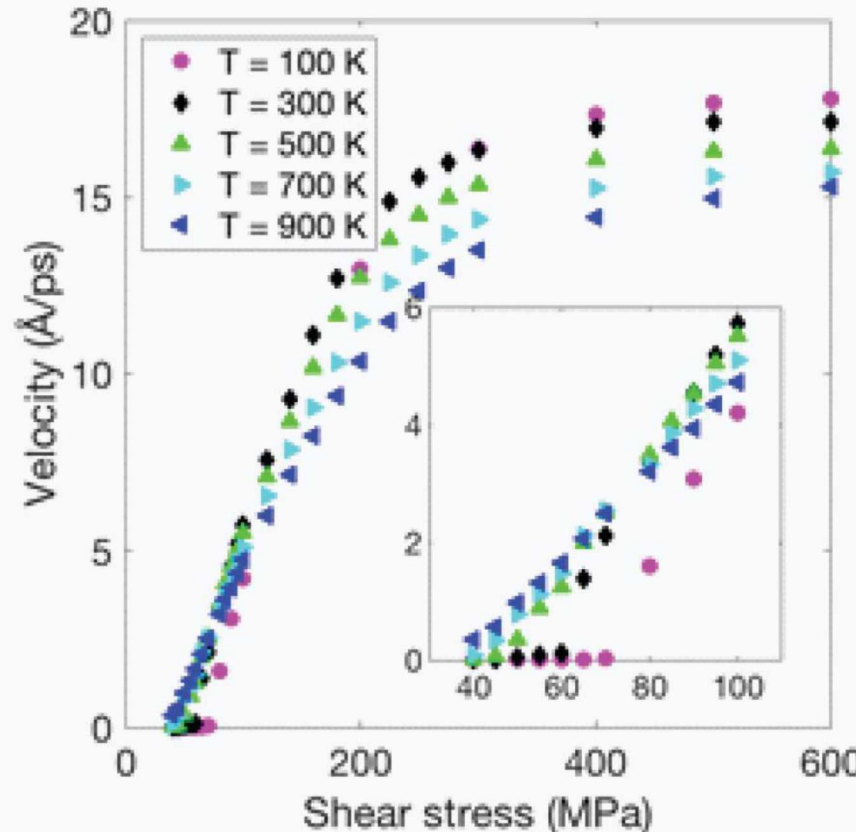
Selected Dislocation Distance/Velocity

Dislocation length $L_z = 3748 \text{ \AA}$

(a) distance at $T = 500 \text{ K}$



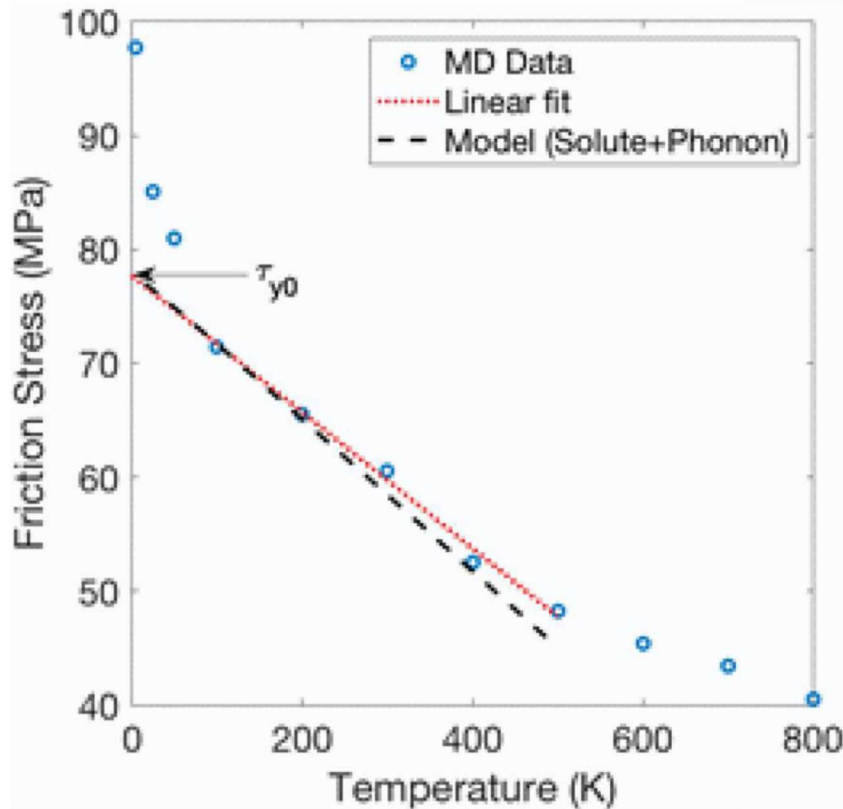
(b) velocity



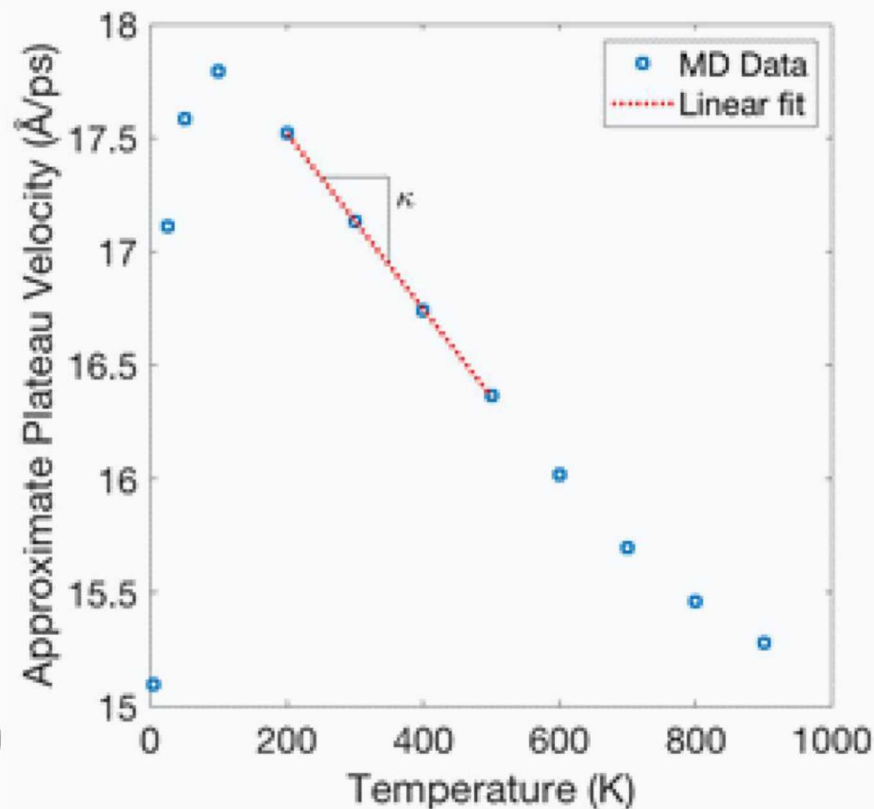
- ❑ Velocities reach plateaus at high stresses.
- ❑ At low stresses, velocities increase with temperature (solute dragging).
- ❑ At high stresses, velocities decrease with temperature (phonon dragging).
- ❑ Fraction stress (τ_f) decreases with temperature.

Friction Stress and Plateau Velocity

(a) friction stress



(b) plateau velocity at $\tau = 600$ MPa



- Friction stress criterion: velocity $\leq 0.25 \text{ \AA}/\text{ps}$.
- At intermediate temperatures, friction stress and plateau velocity both linearly decrease with temperature.

Analytical Mobility Law

Account for solute, phonon, and “singular” drag mechanisms:

$$\tau_d = \frac{1}{b} (F_{sol} + F_{ph} + F_{sing})$$

$$F_{ph} = CT^n v$$

$$F_{sing} = s(c_T - v)^\beta \quad \text{Marian and Caro, 2004.}$$

$$v = 2w\omega_H \exp\left(-\frac{\Delta E_b}{k_B T}\right) \sinh\left(\frac{\Delta E_b}{k_B T} \frac{\tau}{\tau_{y0}}\right)$$

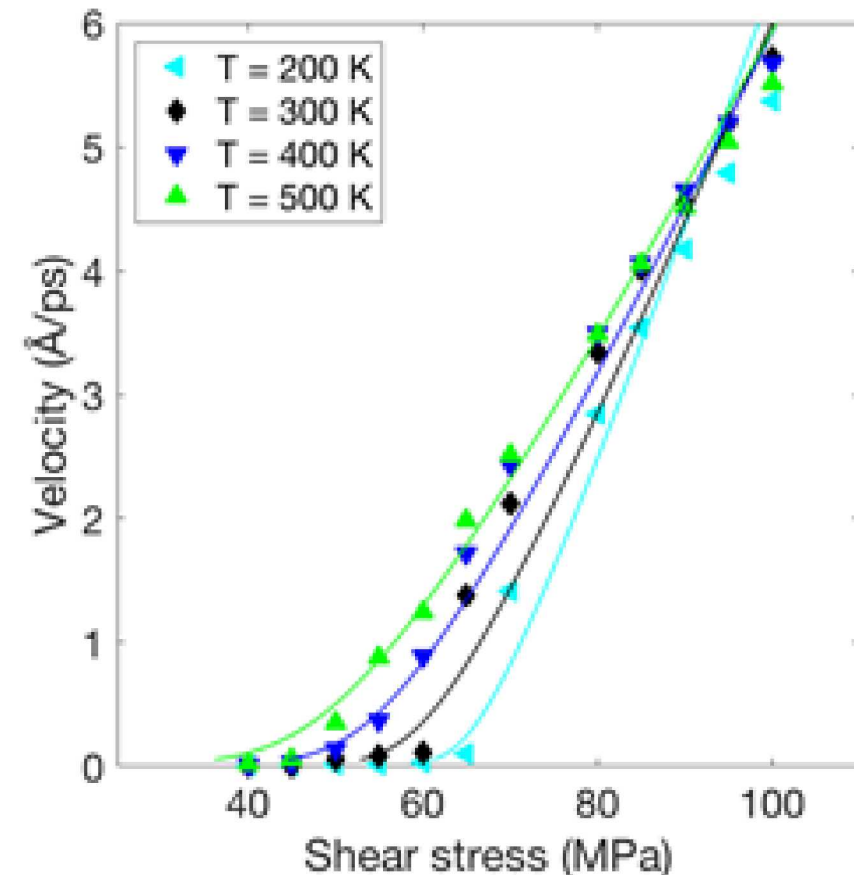
Parameter: ΔE_b , $w\omega_H$, τ_{y0} , C , n , s , c_T , β .

Solute + Phonon only

$$\tau_{y0} = 77.7 \text{ MPa}$$

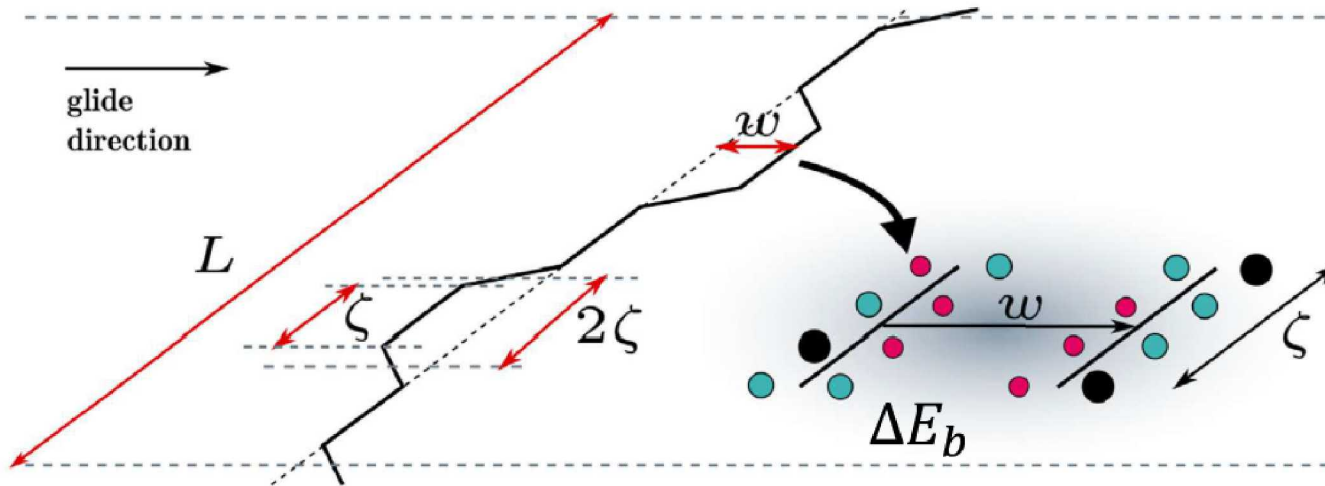
$$\Delta E_b = 0.725 \text{ eV}$$

$$w\omega_b = 4.35 \times 10^4 \text{ m/s}$$

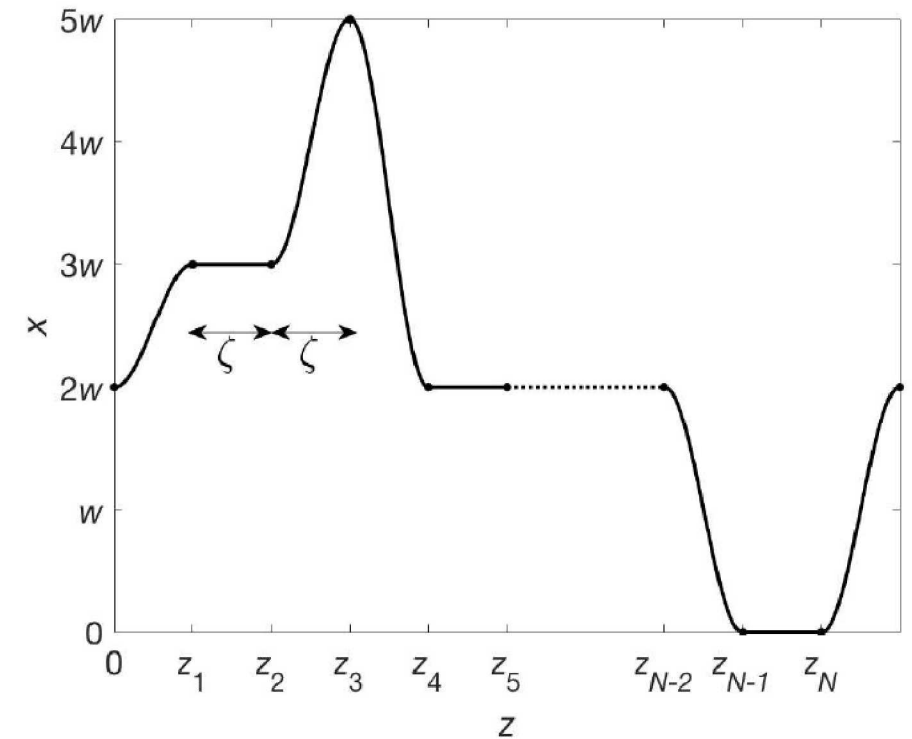


Kinetic Monte Carlo (kMC) Model

Varvenne et al. (2016)



Solid solution strengthening theory of Leyson et al. (2010)

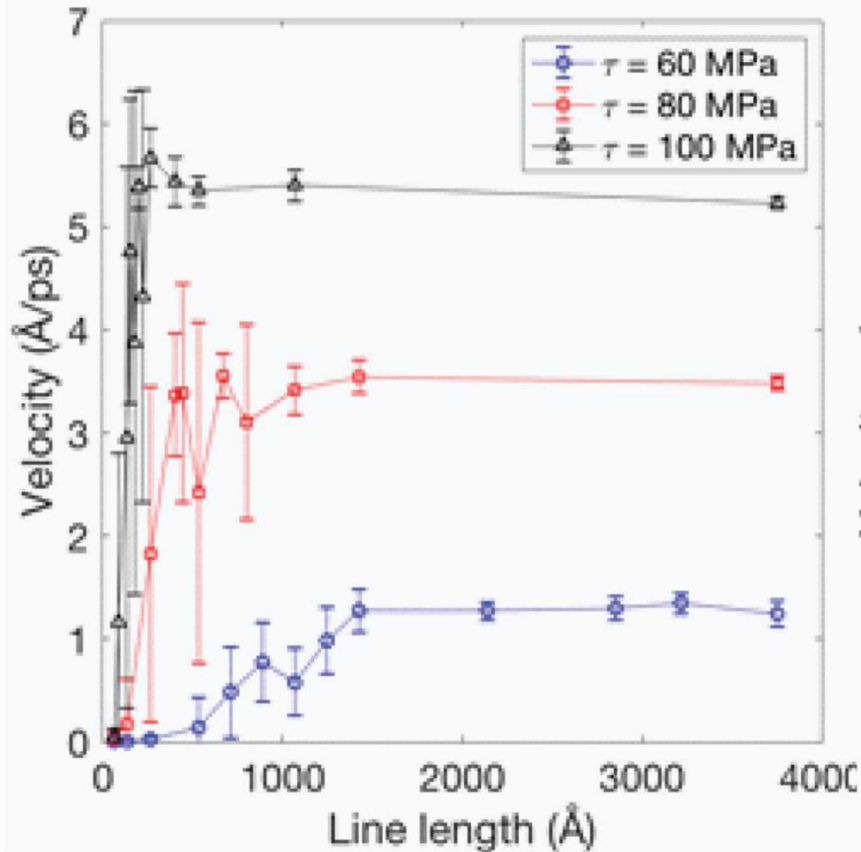


Sinusoidal line profile with discrete “control points”

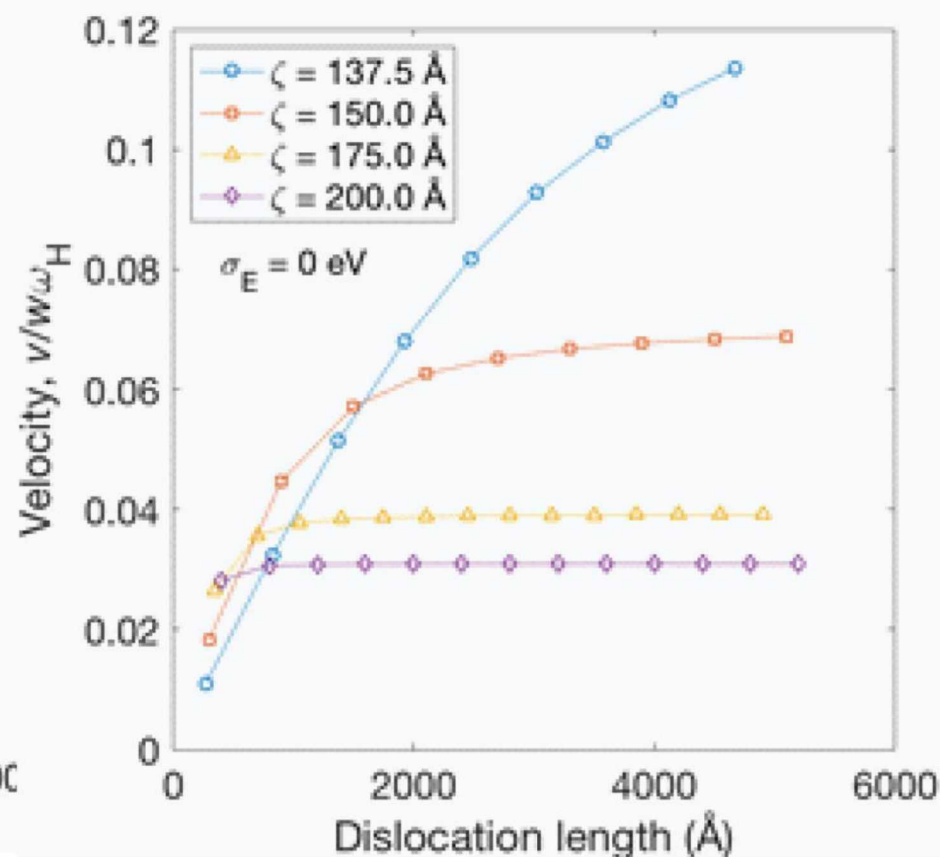
Dislocation migrates through activation of ζ segments.

MD vs. Model on Dislocation Velocity

(a) MD results at 500 K



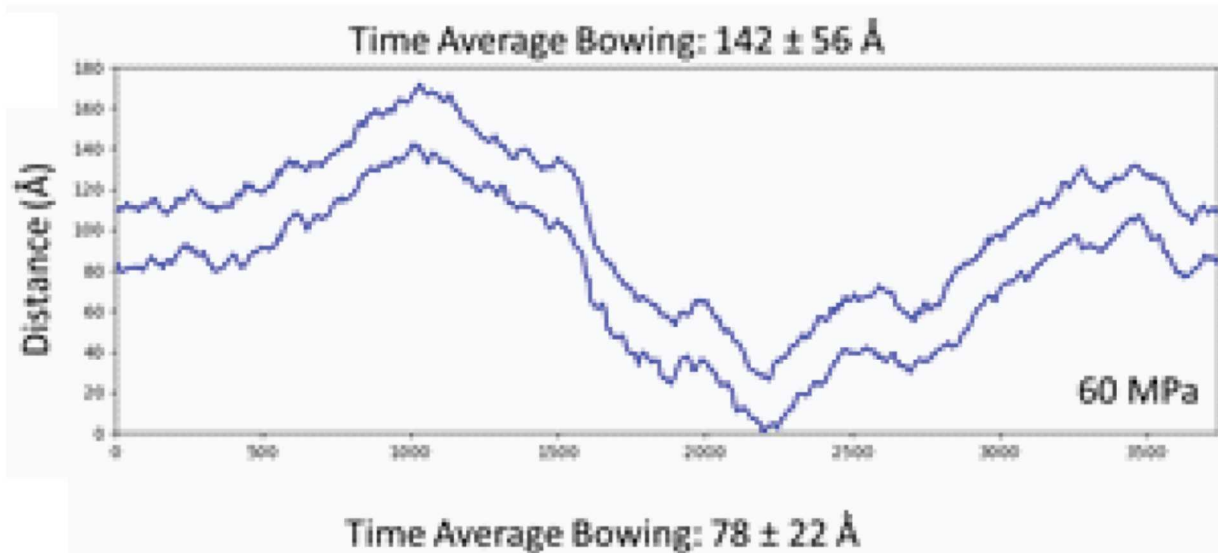
(b) kMC results at 500 K



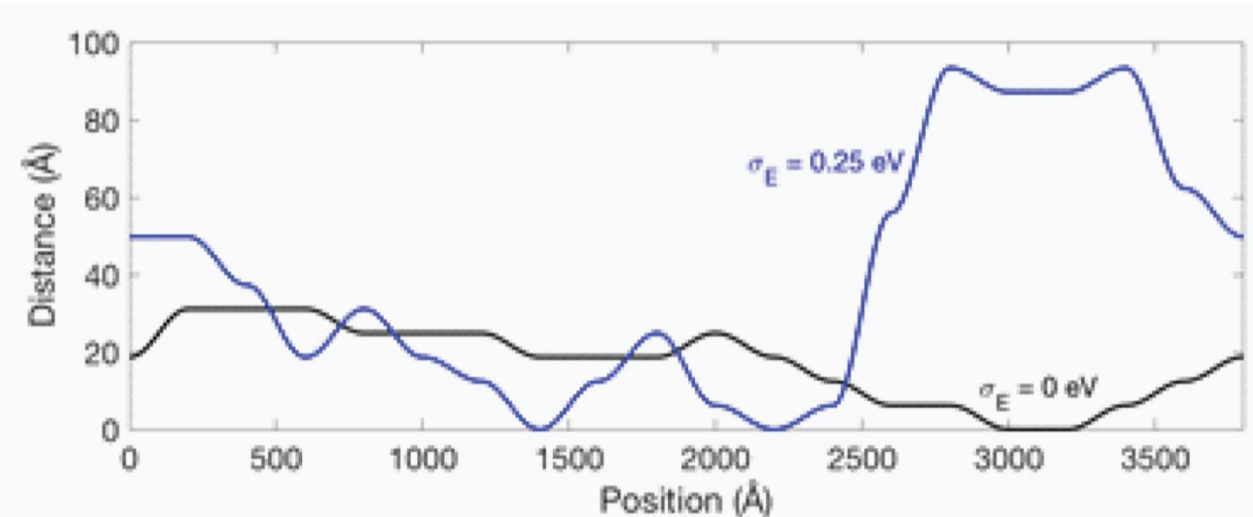
- σ_E : standard deviation of energy barriers of segments ζ .
- Length dependence comes from solute dragging effects.
- Segment length ζ impacts the length dependence.

MD vs. Model on Dislocation Shape

(a) MD results at 500 K, 60 MPa



(b) kMC results at 500 K, 60 MPa, $\zeta = 200 \text{ \AA}$



- ❑ σ_E is standard deviation of energy barrier distribution.
- ❑ Need a spatially varying energy barrier to reproduce large-scale bowing.

Conclusions

- ❑ MD simulations reveal fundamental relationships between stress, temperature, and size.
- ❑ Model accounting for solute, phonon, and singular dragging can rationalize MD results.
- ❑ kMC results indicate that the size dependence originates from activation of local dislocation segments.
- ❑ Energy barrier dislocation impact bowing scale.
- ❑ Results stimulate new strengthening ideas to limit activated dislocation length such as nanostructuring and composition modulation etc.