

# Large-Scale Molecular Dynamics (MD) Derivation of Length-<sub>SAND2019-13245C</sub> Insensitive Dislocation Mobility Laws in Fe-Ni-Cr Stainless Steels

## International Conference on Plasticity, Damage & Fracture

Jan. 3-9, 2020, Rivera Maya, Mexico



X. W. Zhou<sup>1</sup>, R. B. Sills<sup>2</sup>, and M. E. Foster<sup>1</sup>

<sup>1</sup>Sandia National Laboratories, Livermore, California 94550, USA

<sup>2</sup>Rutgers University, Piscataway, New Jersey 08854, USA

Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA-0003525. The authors gratefully acknowledge research support from the U.S. Department of Energy, Office of Energy Efficiency and Renewable Energy, Fuel Cell Technologies Office. The views expressed in the article do not necessarily represent the views of the U.S. Department of Energy or the United States Government.

# Motivation

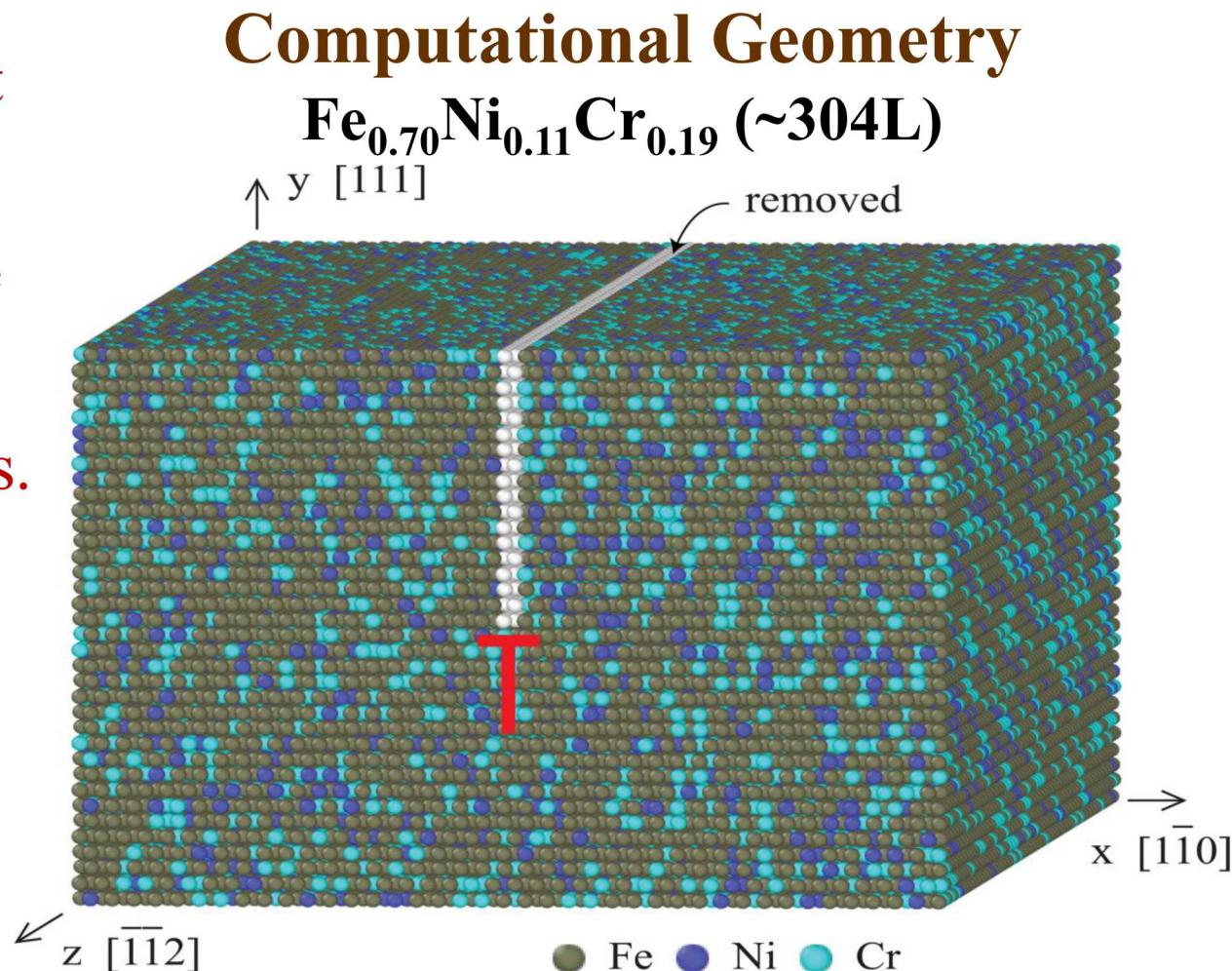
- Dislocation mobility laws are required for dislocation dynamics simulations but cannot be measured experimentally.
- Dislocation mobility in alloys is indicative of solute strengthening.
- Stainless steels are important structural materials.
- Past MD results of dislocation mobility in alloys are size sensitive.
- Mobility laws in a wide range of stresses and temperatures have not been well understood.
- **We attempt to understand size-insensitive mobility laws in a wide stress-temperature space for stainless steels.**

# Goals

- Use MD to systematically study edge dislocation mobility in random  $\text{Fe}_{0.7}\text{Ni}_{0.11}\text{Cr}_{0.19}$  over a range of stresses and temperatures.
- Develop a theoretical mobility model to rationalize the MD data.
- Understand when and why the mobility becomes length-dependent.

# Approach

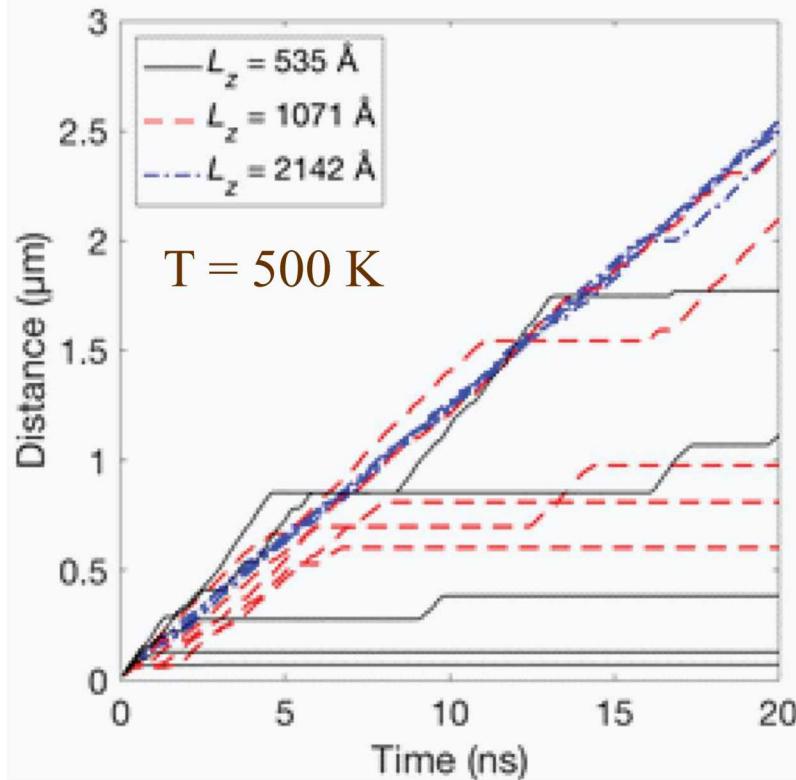
- Use an Fe-Ni-Cr EAM potential (Zhou et al, 2018).
- Stresses are modeled as forces on surface layers of atoms (in  $\pm x$ ).
- NPT simulations are performed for  $\geq 8$  ns.
- Dislocation distance is determined by scaling the relative shift between upper / lower halves of the crystal.
- For each stress / temperature condition, various sizes are explored.



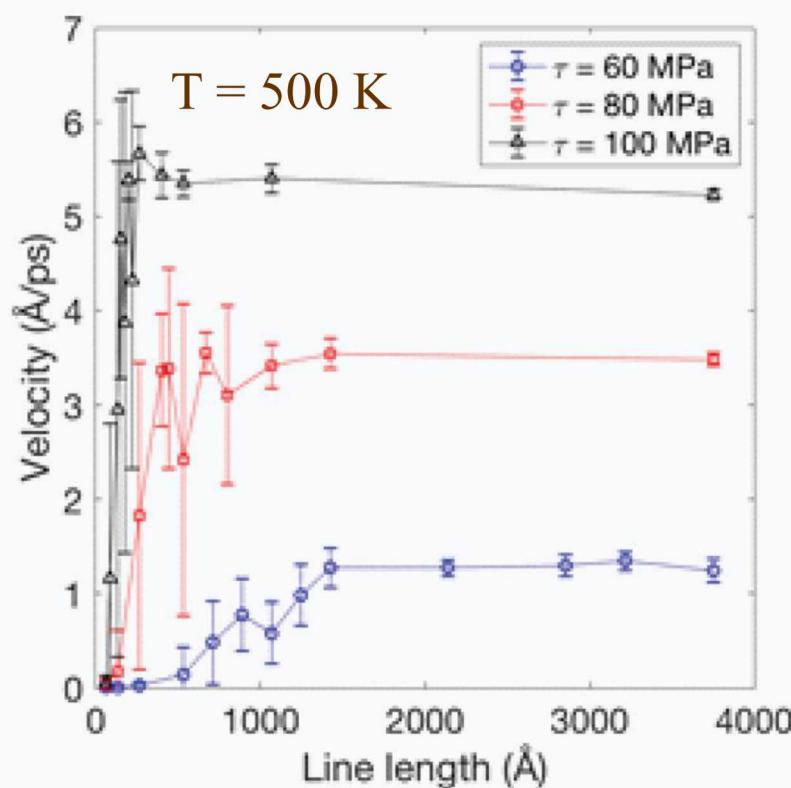
$L_z$ : dislocation length,  $L_y$ : dislocation-surface spacing,  $L_x$ : periodic moving distance.

# Size Effects

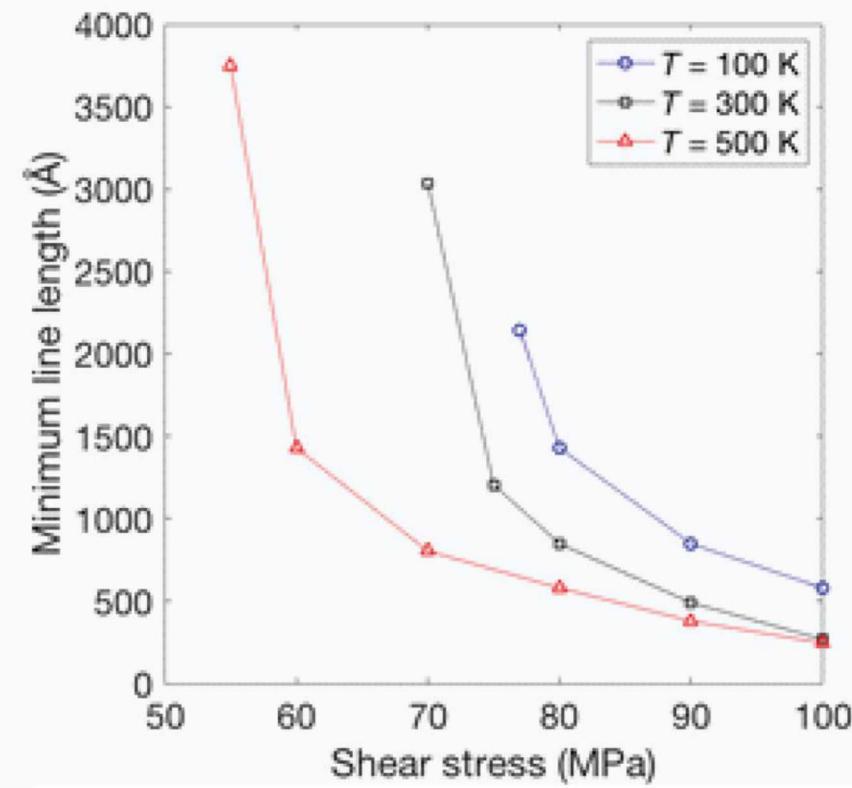
(a) distance at  $\tau = 60$  MPa



(b) velocity



(c) threshold size-independent length

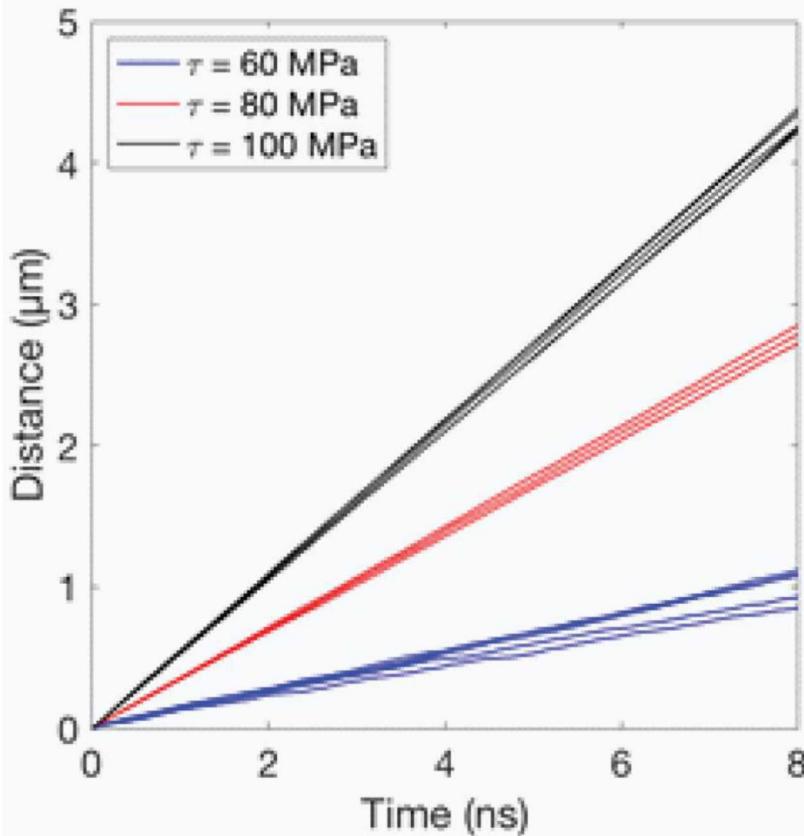


- Velocities are sensitive to dislocation length  $L_z$  but not to  $L_y$  and  $L_x$ .
- Size-insensitive velocities can be obtained at high stresses and temperatures.

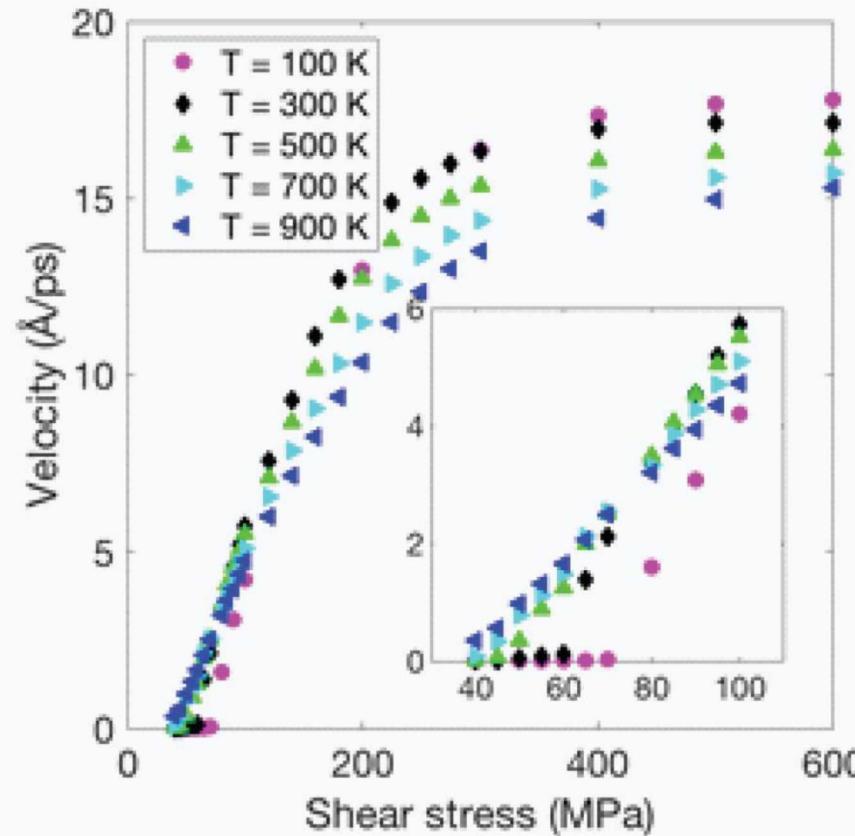
# Selected Dislocation Distance/Velocity

Dislocation length  $L_z = 3748 \text{ \AA}$

(a) distance at  $T = 500 \text{ K}$



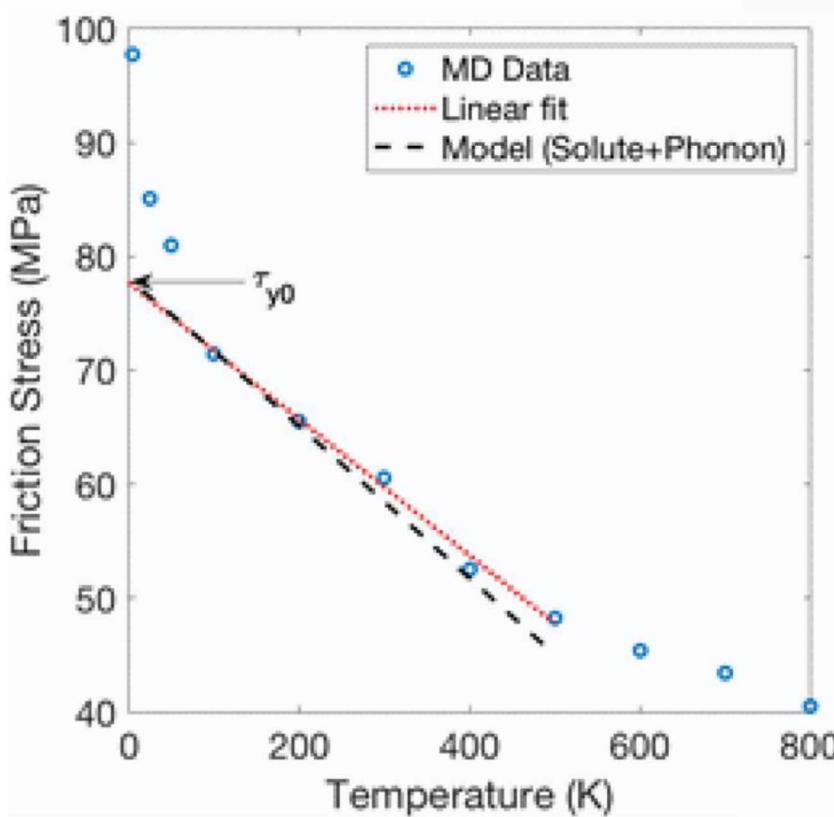
(b) velocity



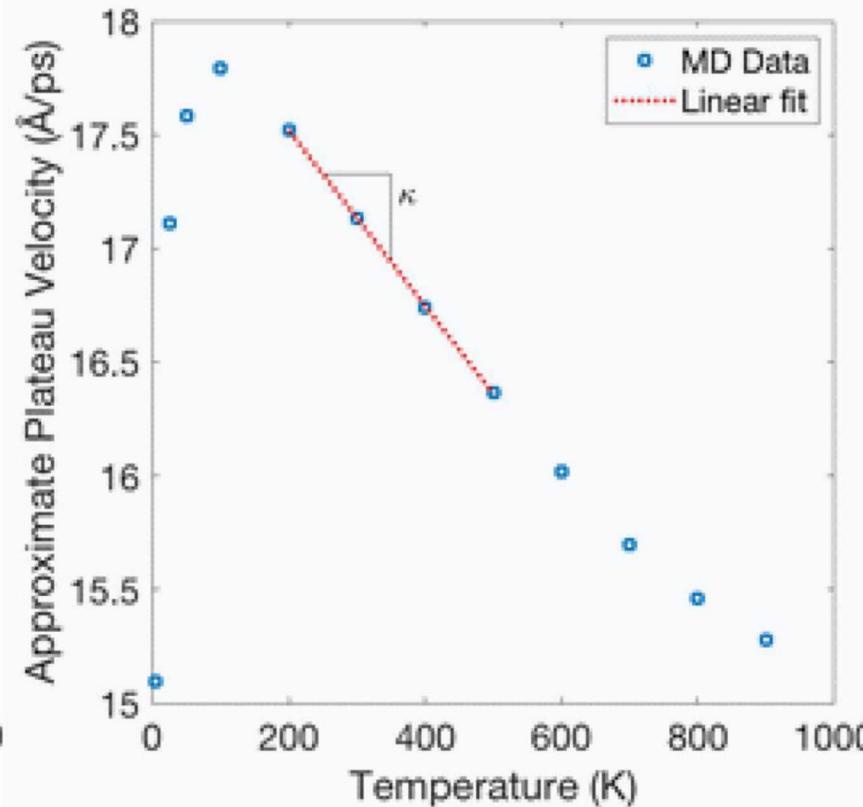
- Velocities reach plateaus at high stresses.
- At low stresses, velocities increase with temperature (solute dragging).
- At high stresses, velocities decrease with temperature (phonon dragging).
- Fraction stress ( $\tau_f$ ) decreases with temperature.

# Friction Stress and Plateau Velocity

(a) friction stress



(b) plateau velocity at  $\tau = 600$  MPa



- Friction stress criterion: velocity  $\leq 0.25 \text{ \AA}/\text{ps}$ .
- At intermediate temperatures, friction stress and plateau velocity both linearly decrease with temperature.

# Analytical Mobility Law

Account for solute, phonon, and “singular” drag mechanisms:

$$\tau_d = \frac{1}{b} (F_{sol} + F_{ph} + F_{sing})$$

$$F_{ph} = CT^n v$$

$$F_{sing} = s(c_T - v)^\beta$$

Marian and Caro, 2004.

$$v = 2w\omega_H \exp\left(-\frac{\Delta E_b}{k_B T}\right) \sinh\left(\frac{\Delta E_b}{k_B T} \frac{\tau}{\tau_{y0}}\right)$$

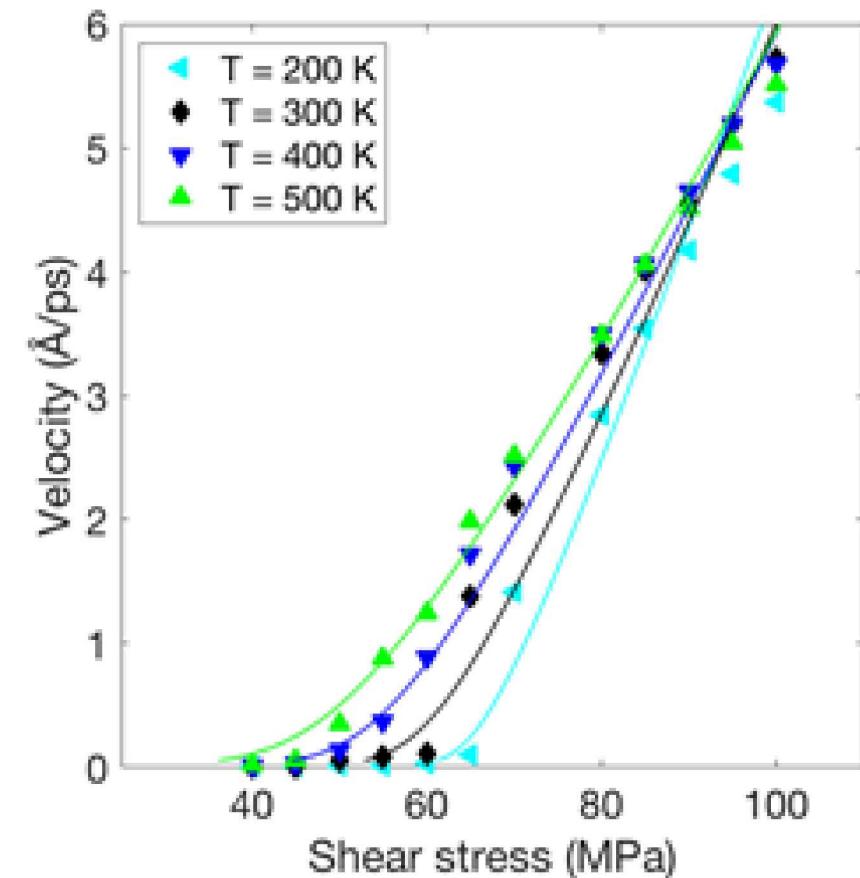
Parameter:  $\Delta E_b$ ,  $w\omega_H$ ,  $\tau_{y0}$ ,  $C$ ,  $n$ ,  $s$ ,  $c_T$ ,  $\beta$ .

Solute + Phonon only

$$\tau_{y0} = 77.7 \text{ MPa}$$

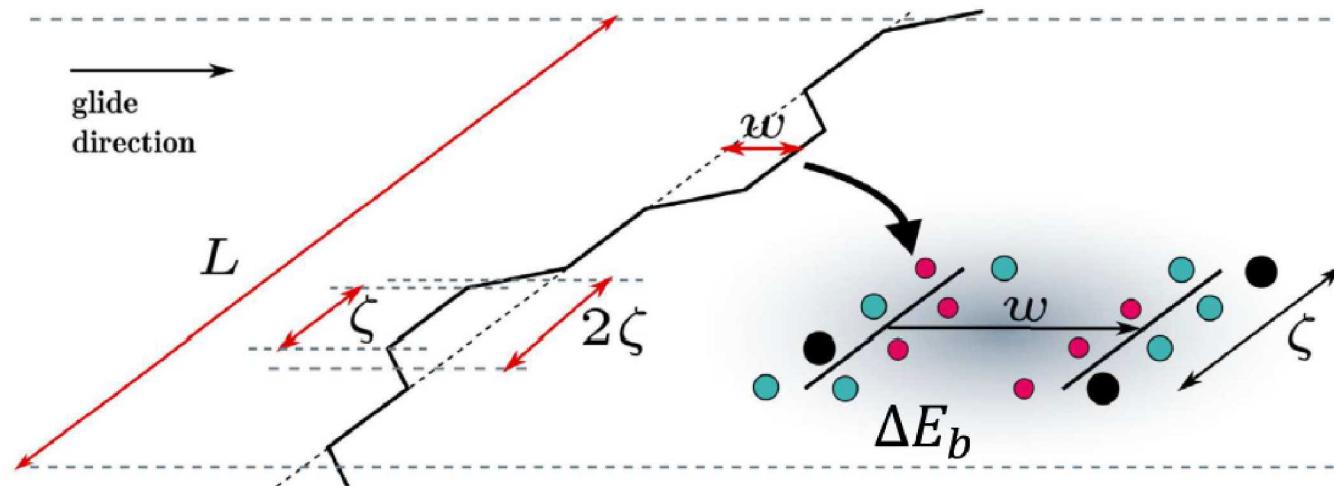
$$\Delta E_b = 0.725 \text{ eV}$$

$$w\omega_b = 4.35 \times 10^4 \text{ m/s}$$



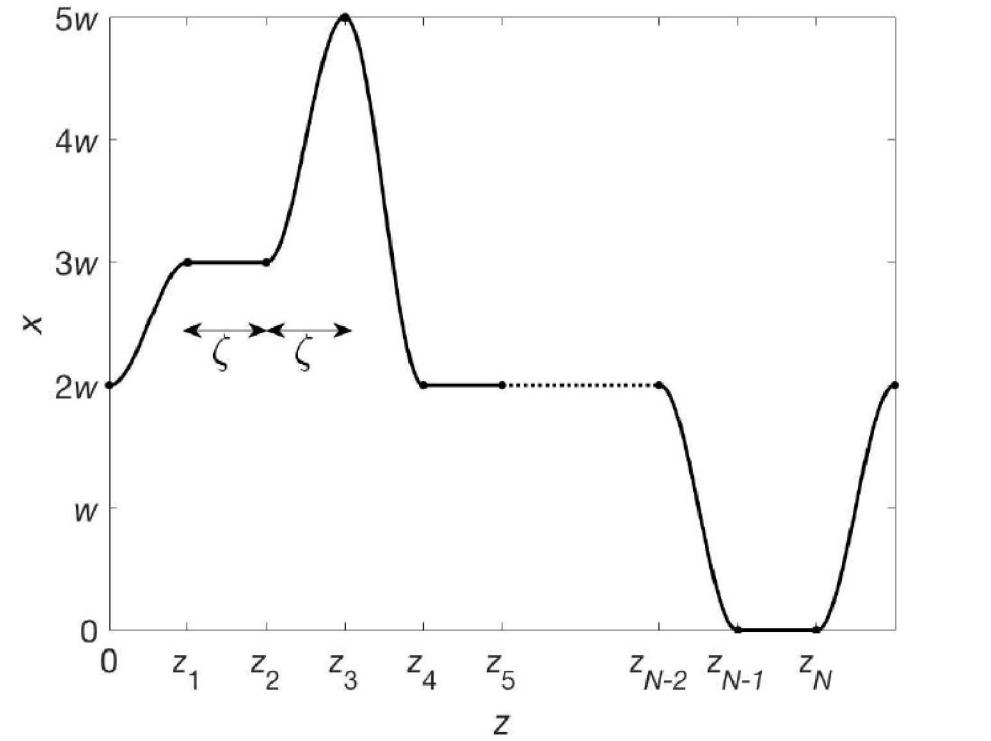
# Kinetic Monte Carlo (kMC) Model

Varvenne et al. (2016)



Solid solution strengthening theory of  
Leyson et al. (2010)

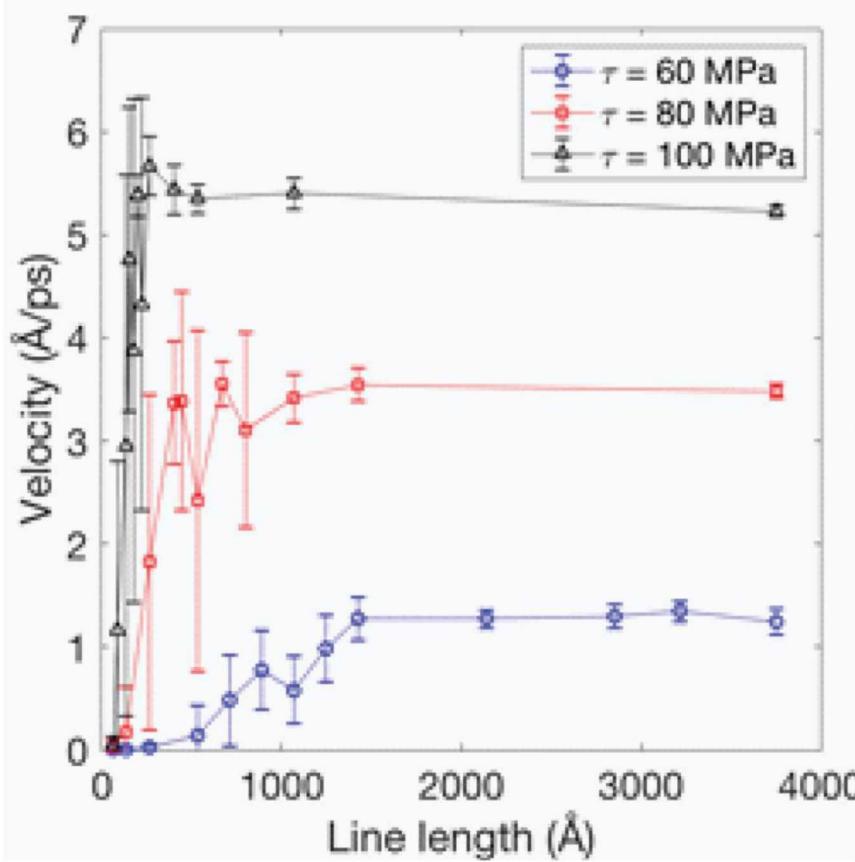
Dislocation migrates through activation of  $\zeta$  segments.



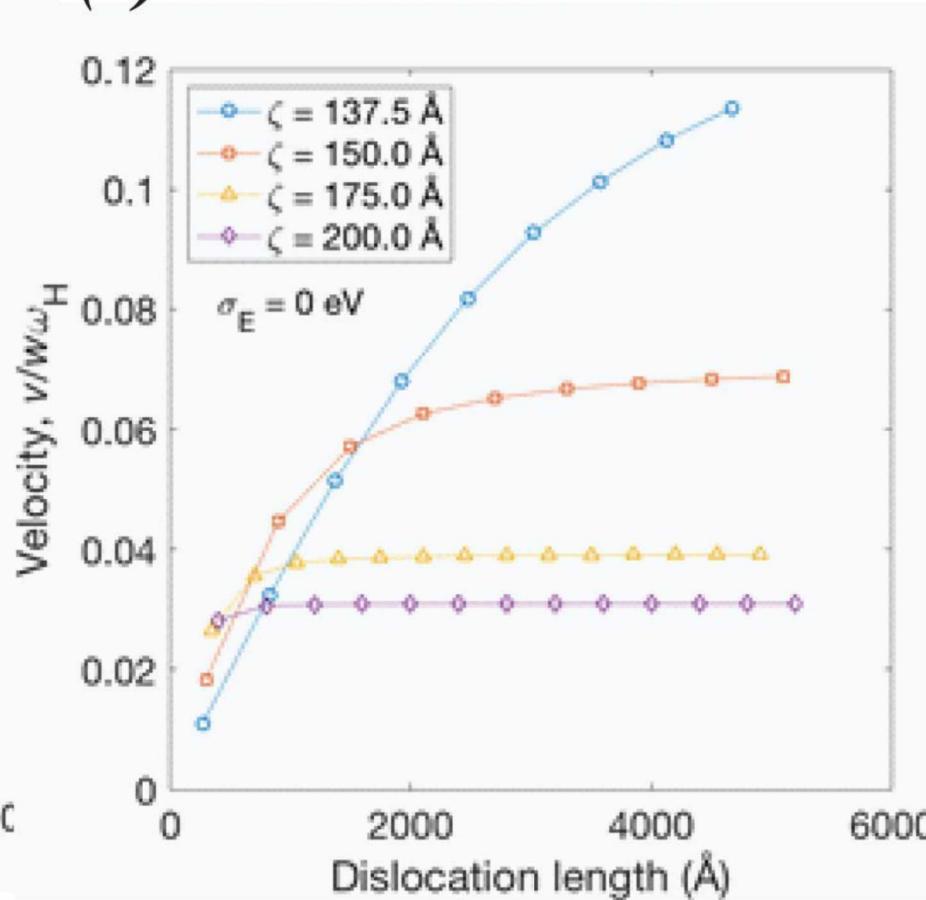
Sinusoidal line profile with discrete  
“control points”

# MD vs. Model on Dislocation Velocity

(a) MD results at 500 K



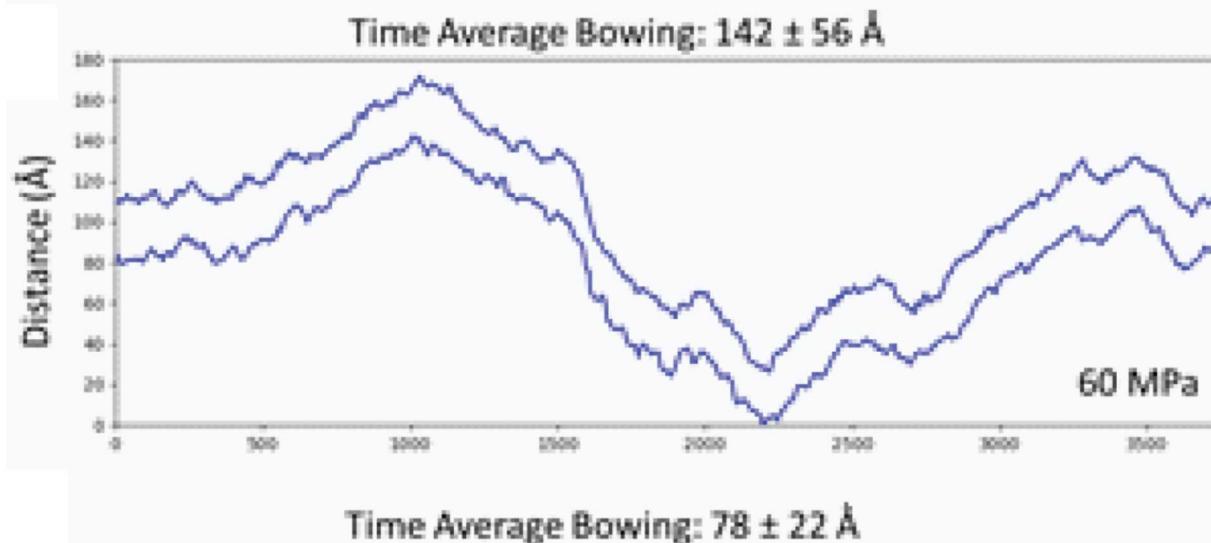
(b) kMC results at 500 K



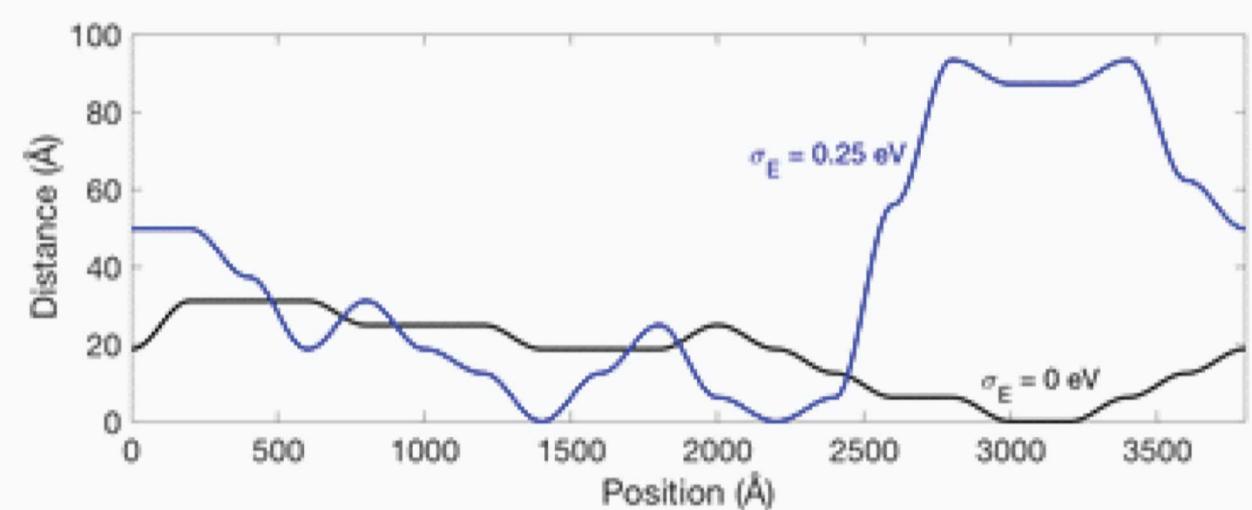
- $\sigma_E$ : standard deviation of energy barriers of segments  $\zeta$ .
- Length dependence comes from solute dragging effects.
- Segment length  $\zeta$  impacts the length dependence.

# MD vs. Model on Dislocation Shape

(a) MD results at 500 K, 60 MPa



(b) kMC results at 500 K, 60 MPa,  $\zeta = 200 \text{ \AA}$



- $\sigma_E$  is standard deviation of energy barrier distribution.
- Need a spatially varying energy barrier to reproduce large-scale bowing.

# Conclusions

- MD simulations reveal fundamental relationships between stress, temperature, and size.
- Model accounting for solute, phonon, and singular dragging can rationalize MD results.
- kMC results indicate that the size dependence originates from activation of local dislocation segments.
- Energy barrier dislocation impact bowing scale.
- Results stimulate new strengthening ideas to limit activated dislocation length such as nanostructuring and composition modulation etc.