



Computation of Embrittling Potencies of Sulfur for a Range of Nickel Pure Tilt Grain Boundaries via Atomistic Simulation Methods

Doruk Aksoy¹, Remi Dingreville² and Douglas E. Spearot¹

¹ Department of Mechanical & Aerospace Engineering
University of Florida, Gainesville, FL

² Sandia National Laboratories, Albuquerque, NM

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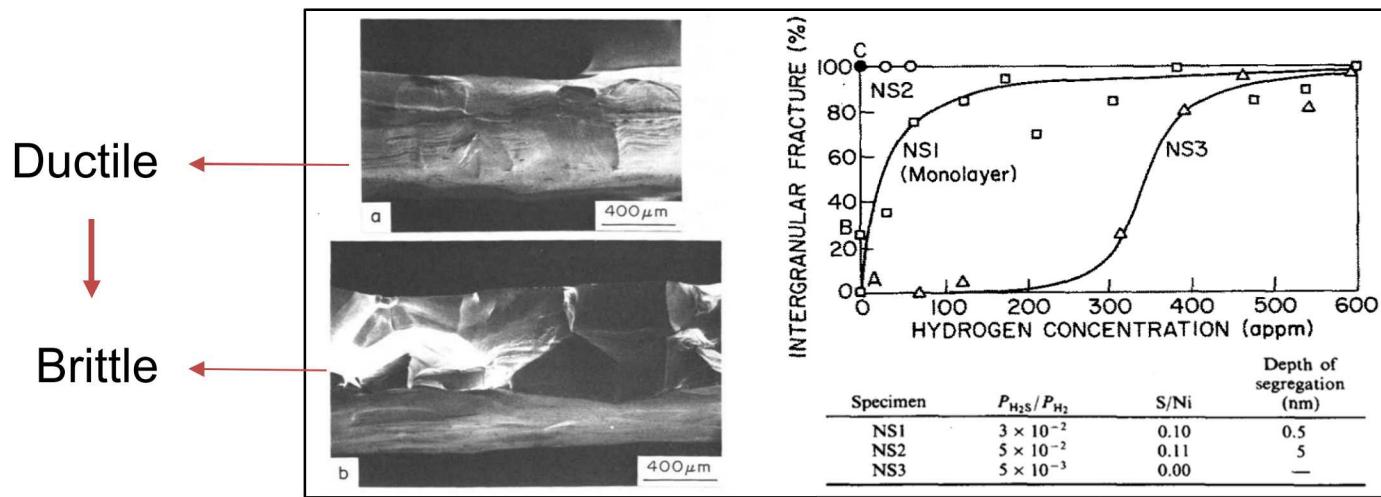
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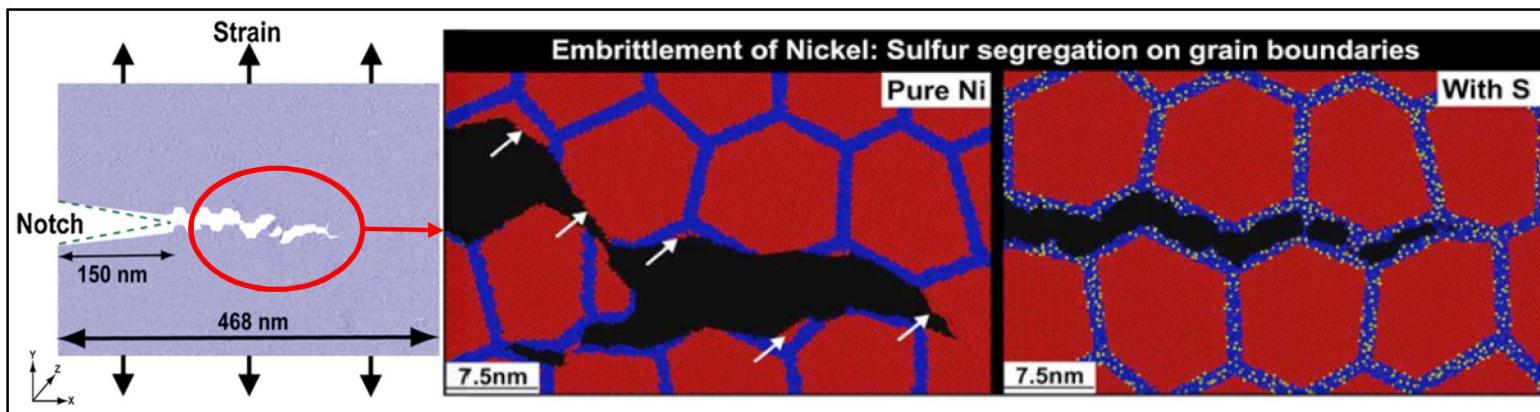


Introduction

- **Embrittlement** is defined as the loss of ductility in metals caused by segregation of certain impurity atoms.



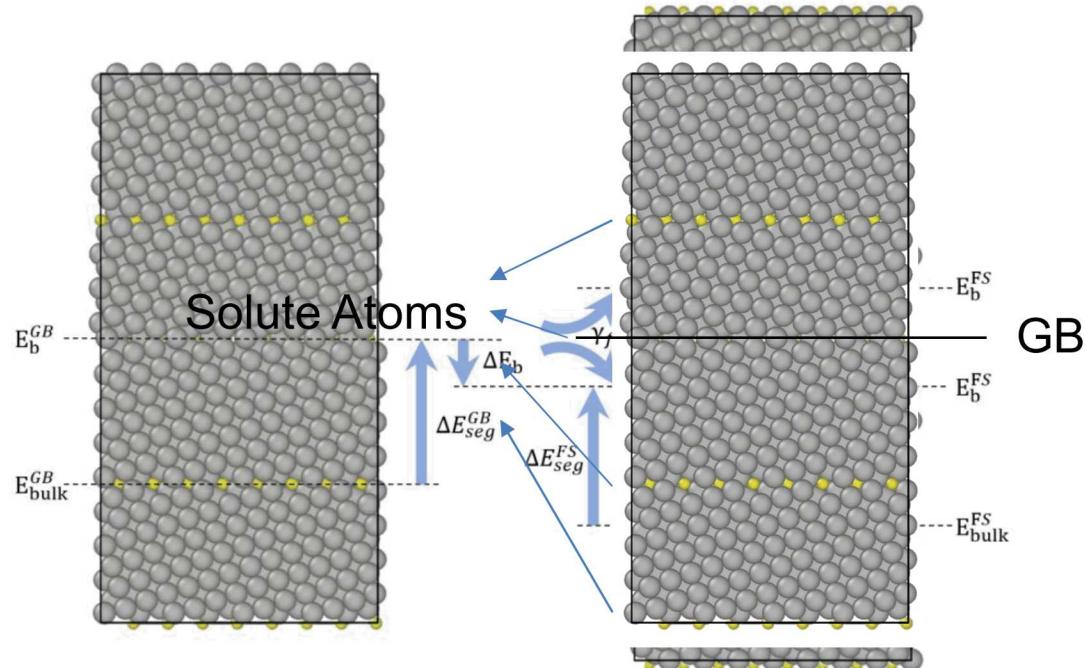
Lassila and Birnbaum (1987)



Ductile

Brittle Chen et al. (2010)

What is Grain Boundary Segregation Induced Embrittlement?



- **Segregation energy** is defined as the binding energy difference between the bulk site and a GB/ FS site.

$$\Delta E_{seg}^{GB/FS} = E_b^{GB/FS} - E_b^{GB/bulk}$$

- **Embrittling potency** is defined as the difference between the GB and FS segregation energies

$$\Delta E_b = \Delta E_{seg}^{GB} - \Delta E_{seg}^{FS}$$

Objectives

Motivation

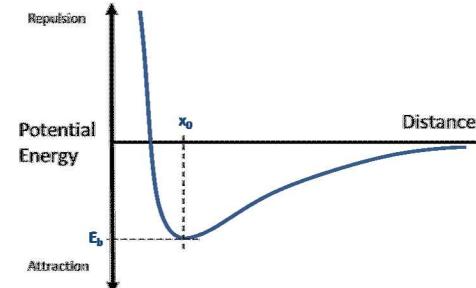
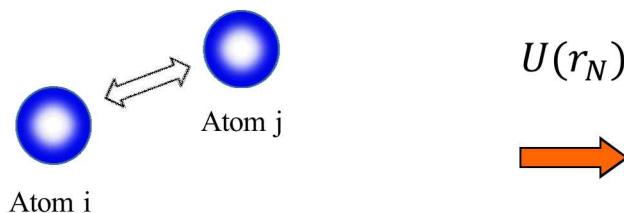
- The effect of GB structure on embrittlement behavior needs explanation.
- A computationally efficient interatomic potential for Ni-S for high-throughput simulations is lacking.

Objectives

- To develop a new Embedded-Atom Method (EAM) interatomic potential **parameterized for S induced embrittlement of Ni** to perform molecular dynamics (MD) simulations
- To calculate segregation energies and embrittling potencies for various STGBs with the newly developed EAM potential

Interatomic Potential

- Interatomic potentials define the interactions between atoms in order to simulate material behaviors such as fracture, phase transformations, deformations etc.



- One of the most computationally efficient interatomic potentials developed for modeling metals and impurities is the **Embedded-Atom Method (EAM)**.
- In EAM, all atoms have an embedding energy which depend on the environmental effects of the host lattice.
- Total energy of the system: $E_{tot} = \sum_i F_i(\rho_i) + \frac{1}{2} \sum_{i,j} \phi_{ij}(r_{ij})$
- 3 main functions needed to fit an EAM potential:
 1. Embedding Energy Function: $F(\rho)$
 2. Host Electron Density Function: $\rho(r)$
 3. Pair Potential Function: $\phi(r)$

Interatomic Potential Development

- In order to develop a new Ni-S EAM potential, the following functions are selected¹ for the fitting procedure:

Host Electron Density Function

$$\rho(r) = \rho_0 e^{-\frac{r-r_0}{\lambda}} \Xi(r, r_d) \quad \text{where} \quad \Xi(\chi) = -6\chi^5 + 15\chi^4 - 10\chi^3 + 1 \quad \text{and} \quad \chi(r, r_c) = \frac{r - r_c}{r_{cut} - r_c} H(r_c)$$

Embedding Energy Function

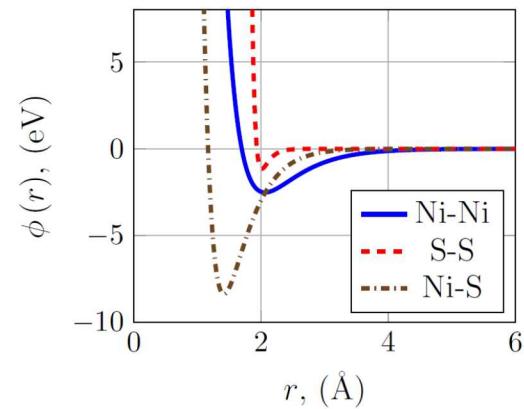
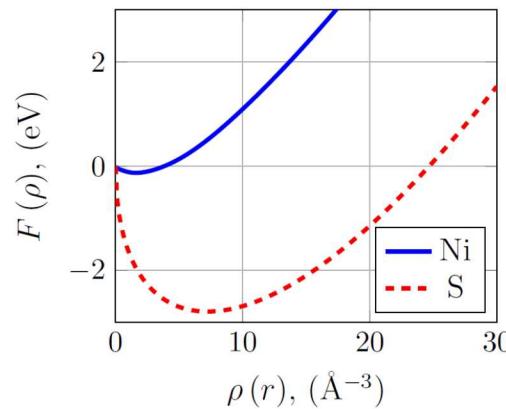
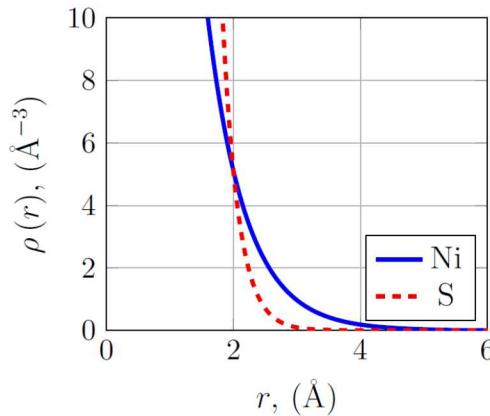
$$E_{tot} = \sum_i F_i(\rho_i) + \frac{1}{2} \sum_{i,j} \phi_{ij}(r_{ij}) \quad \text{where} \quad \rho_i = \sum_{j \neq i} \rho(r_{ij})$$

Pair Potential Function

$$\phi(r) = D_e \{ [1 - e^{-a_m(r-r_0)}]^2 - 1 \} \cdot \Xi(r, r_p)$$

- 3 fitting procedure needed to describe Ni-S interactions:

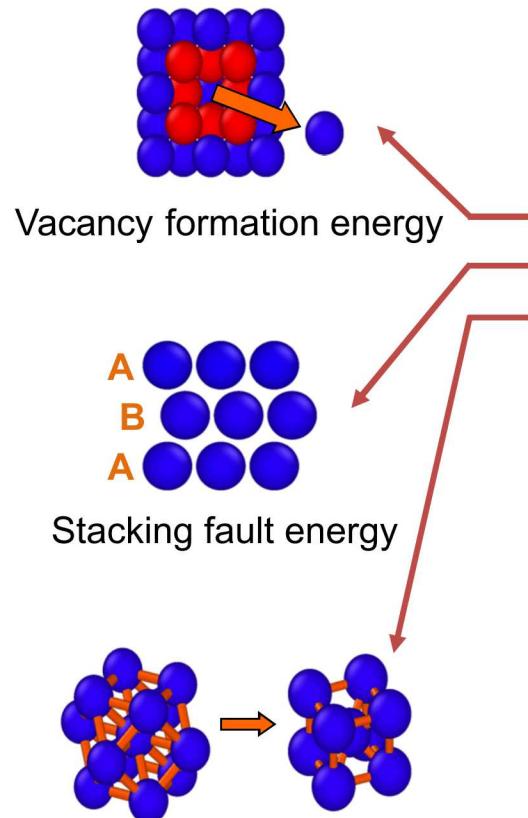
➤ Ni-Ni, S-S and Ni-S



¹ O'Brien *et al.* (2018) J. Mater. Sci.

Interatomic Potential for Ni-Ni interactions

Target properties and corresponding fit results:



Property ^a	Target ^b	Fit	Experimental ^c
a_0	3.52	3.52	3.52
E_{coh}	-4.45	-4.45	-4.45
E_{vf} *	1.592	1.592	1.60
E_{sf} *	88.843	88.843	125
$\Delta E_{fcc-bcc}$ *	0.077	0.077	0.06
C_{11} *	2.467	2.455	2.465
C_{12} *	1.475	1.491	1.473
C_{44} *	1.250	1.343	1.247

^a a_0 (Å) is the lattice parameter. E_{coh} (eV), E_{vf} (eV) and E_{sf} (eV) are the cohesive, vacancy formation and stacking fault energies, respectively. $\Delta E_{fcc-bcc}$ (eV) is the energy difference between FCC and BCC lattices. C_{11} , C_{12} , C_{44} are the elastic constants with units 10^{12}dyn/cm^2 .

^b Values obtained from MD simulations using Ni-Al-H EAM potential Angelo and Baskes (1995, 1997).

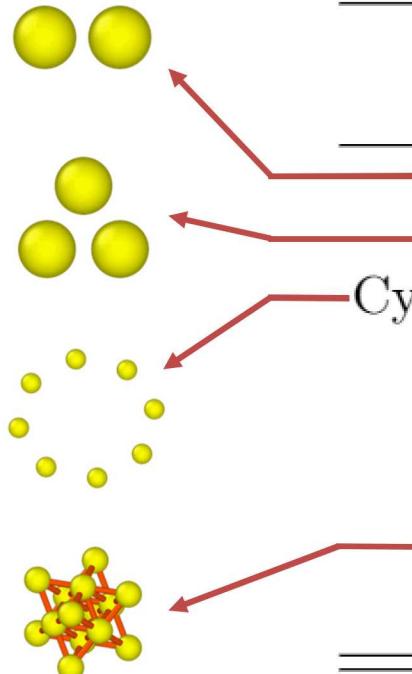
^c Experimental values taken from Angelo and Baskes (1995, 1997).

The energy difference between FCC and BCC state of Ni

Target energies for Ni-Ni interactions are matched exactly

Interatomic Potential for S-S interactions

Target properties and corresponding fit results:



Structure	Target ^a a_0 (Å) ^b	Target ^a E_0 (eV) ^b	Fit a_0 (Å) ^b	Fit E_0 (eV) ^b
Dimer *	1.91	-3.28	2.00	-3.28
Trimer *	2.03	-3.86	2.03	-3.86
Cyclo-octasulfur *	2.04	-4.06	2.03	-3.86
Hexagonal	2.32	-3.30	2.15	-3.45
Diamond	5.79	-3.13	5.24	-3.22
SC	2.59	-3.51	2.35	-3.11
FCC *	4.00	-2.84	3.66	-2.84
BCC	3.18	-3.01	2.89	-2.88

^a Target values are taken from the DFT database¹.

^b a_0 (Å) is the lattice parameter and E_0 (eV) is the energy per atom.

Target energies for S-S structures are in good agreement

Interatomic Potential for Ni-S interactions

Target properties and corresponding fit results:

$\Sigma 5 [100]/(210) 36.87^\circ$

Property ^a	Target ^b	Fit	ReaxFF ^c
ΔE_{seg}^{GB} *	-1.40	-1.40	-1.34
ΔE_{seg}^{FS} *	-2.39	-2.69	-2.96
E_{bulk}^{GB} *	-3.22	-3.22	-3.37
E_{bulk}^{FS}	-	-3.22	-3.35
E_b^{GB}	-	-4.62	-4.71
E_b^{FS}	-	-5.91	-6.31
ΔE_b	0.99	1.29	1.62

^a All energies are in (eV). $\Delta E_{seg}^{GB/FS}$ is the segregation energy associated with a GB/FS cell, $E_{bulk}^{GB/FS}$ is the bulk energy associated with a GB/FS cell, $E_b^{GB/FS}$ is the binding energy associated with a GB/FS cell and ΔE_b is the embrittling potency.

^b Target values are taken from Všianská *et al.* (2011).

Pair interactions for Ni-S is more favorable than S-S interactions, causing the attraction between S-S to diminish

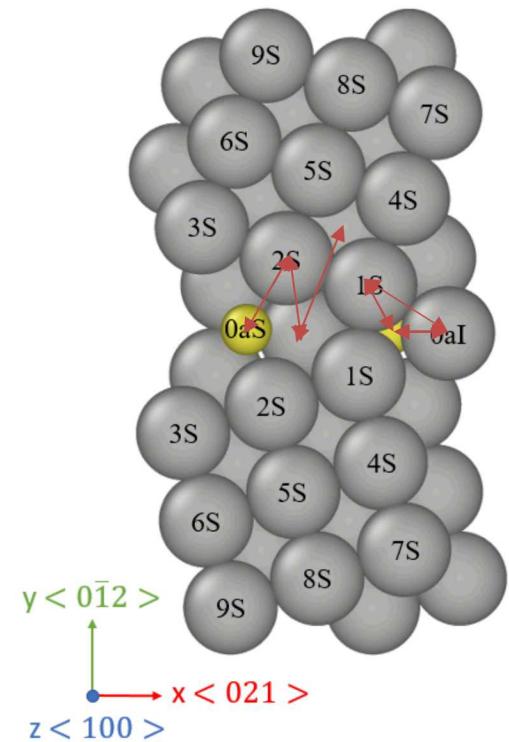
Results not Included in the Fitting Database

- Ni-S bond lengths less than 3 Å are computed for a $\Sigma 5$ STGB with S atoms at the substitutional GB site: “0aS”

$\Sigma 5 [100]/(210) 36.87^\circ$

	Present Work		Yamaguchi <i>et al.</i> (2015) DFT calculations	
	0aS (S)	0aI (Ni)	0aS (S)	0aI (Ni)
0aS (S)	-	2.28	-	2.25
1S (Ni)	2.26	2.64	2.27	2.65
2S (Ni)	2.18	2.74	2.17	2.76
3S (Ni)	-	2.84	-	2.87

* All lengths are in Angstroms.

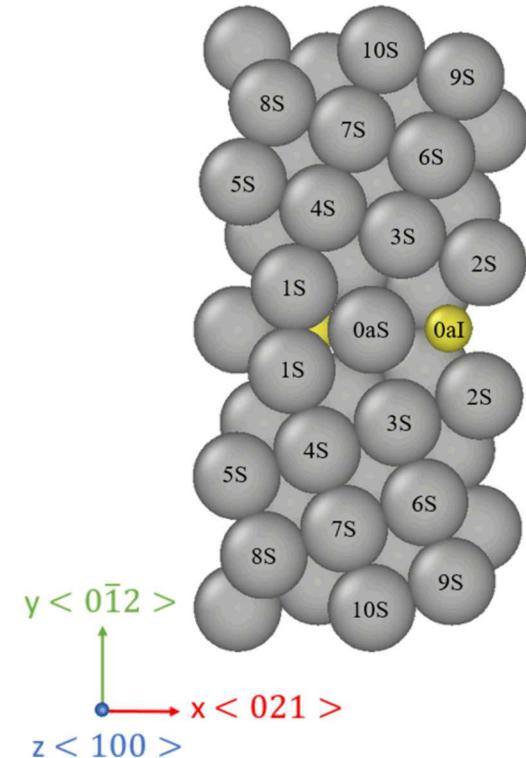
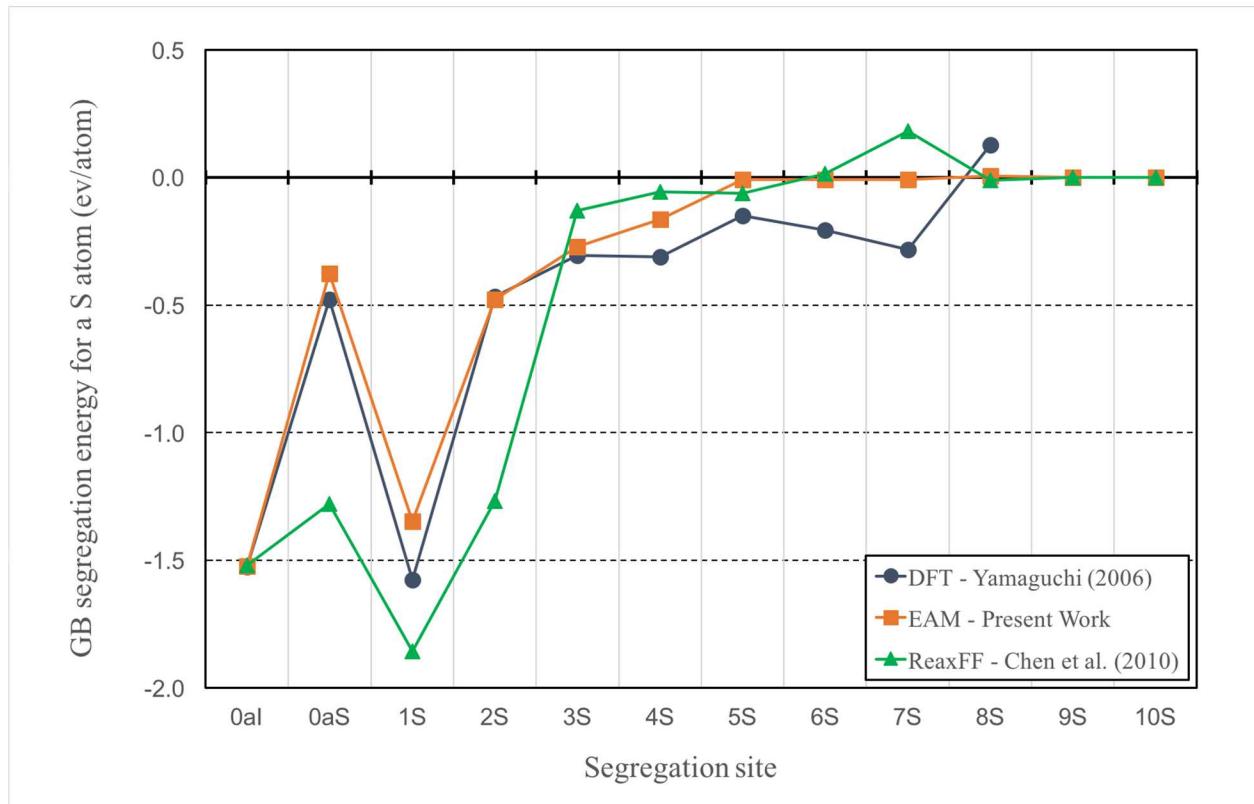


Ni-S bond lengths calculated by our EAM potential are in good agreement with DFT results.

Results not Included in the Fitting Database

- Assumption: S prefers interstitial sites at GB/FS region and substitutional sites at bulk region

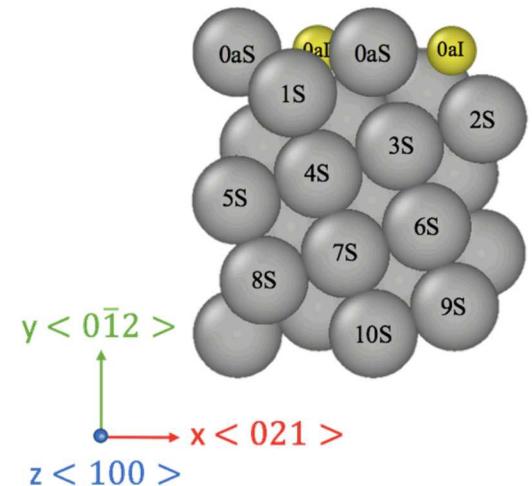
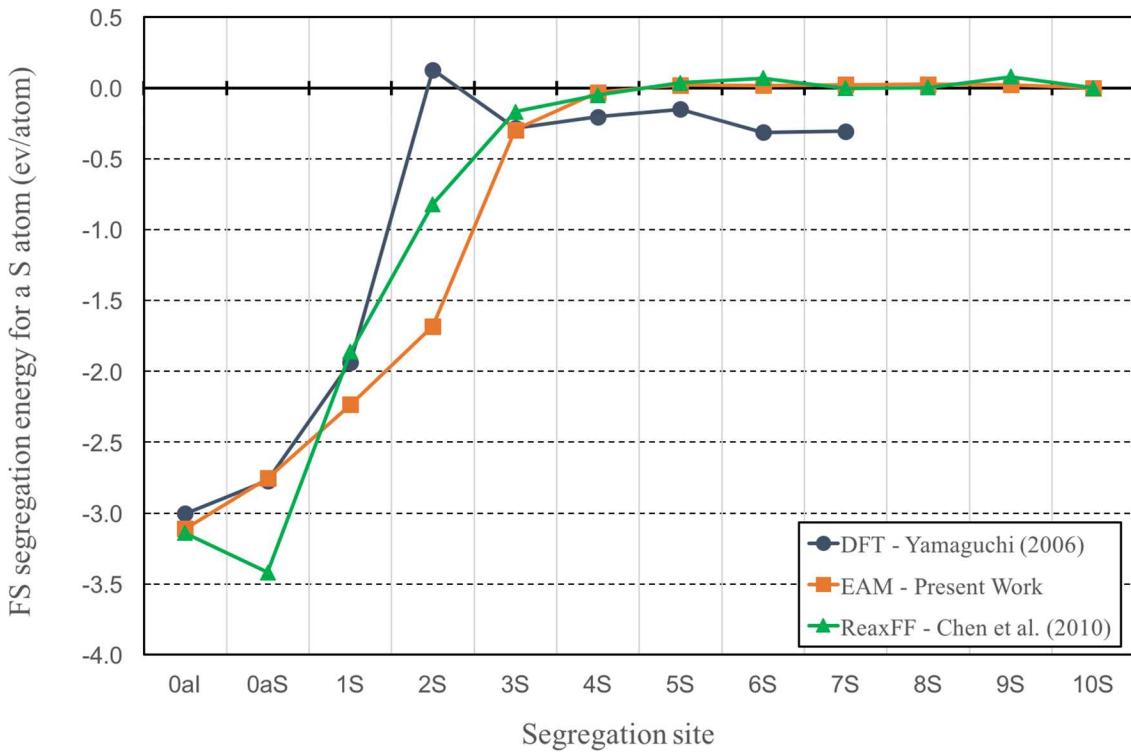
$\Sigma 5 [100]/(210) 36.87^\circ$



The segregation energies associated with different segregation sites are matching the DFT results

Results not Included in the Fitting Database

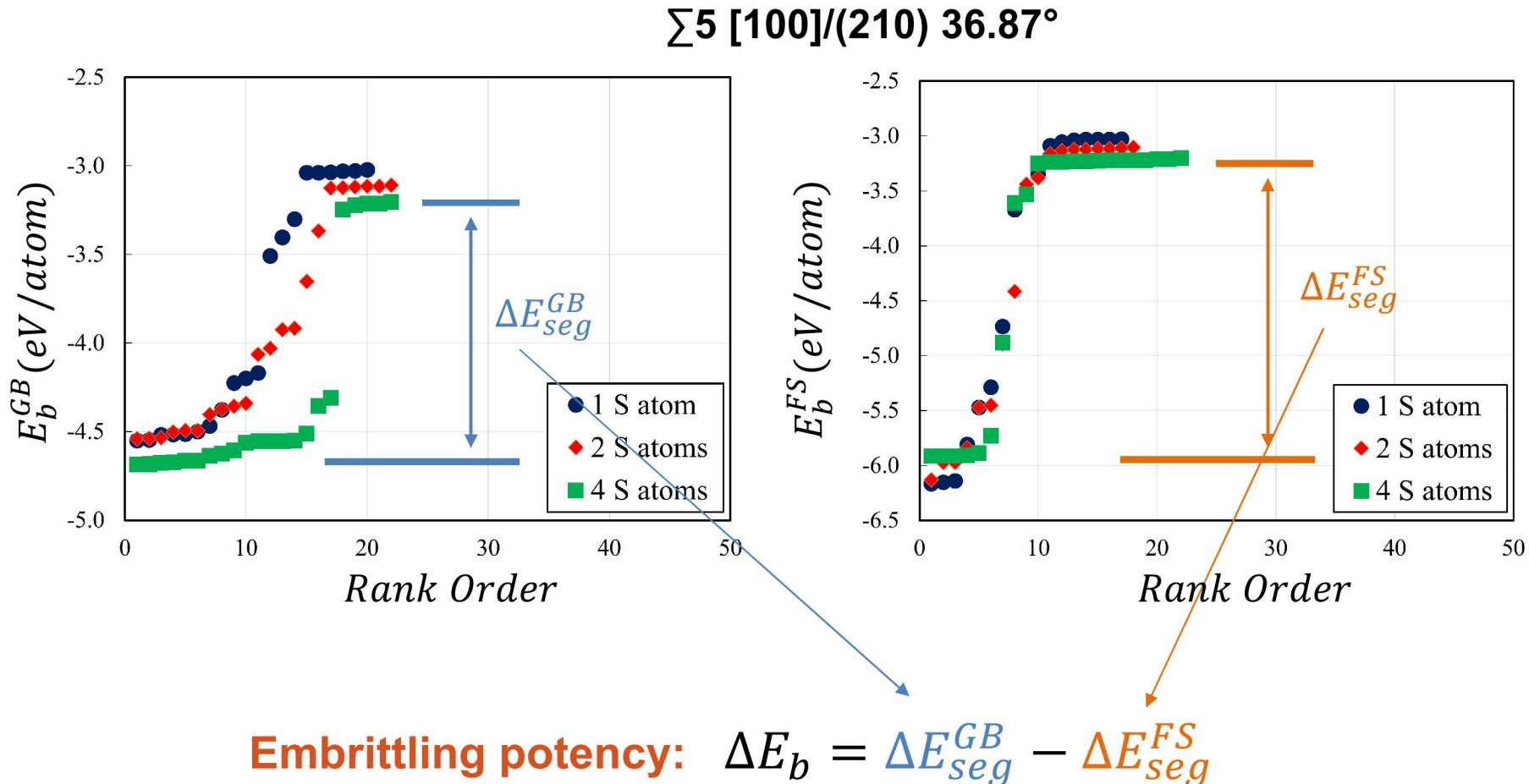
$$\Sigma 5 [100]/(210) 36.87^\circ$$



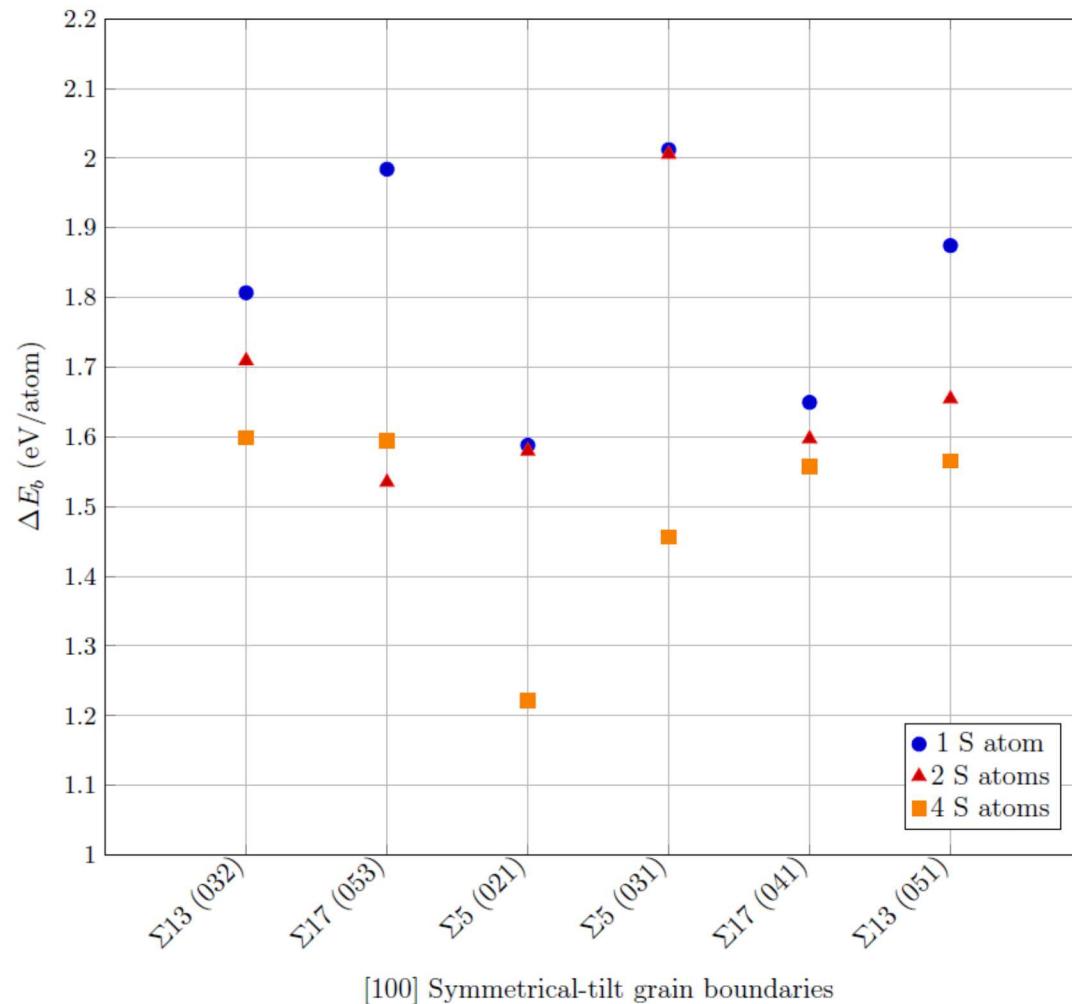
The segregation energies associated with different segregation sites are matching the DFT results

Effect of S Coverage

- Rank ordered binding and segregation energies obtained from the newly developed EAM potential shows the effect of S coverage in the system.



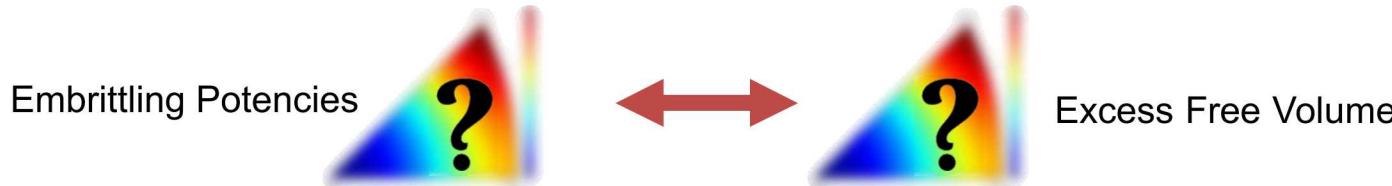
Embrittling Potencies for six [100] STGBs



The embrittling potency of S varies
for different GBs as well as different S coverages.

Systematic Comparison and Future Directions

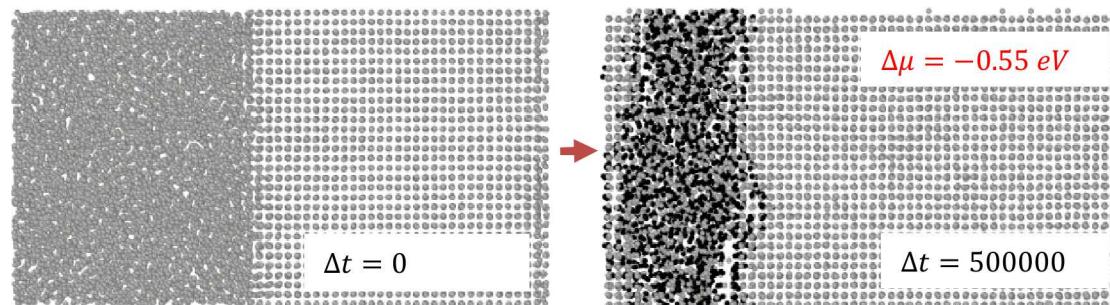
- Same type of approach can be used to plot embrittling potencies for a wide range of GBs, and visualized using stereographic projection.



- Impurity atoms can be introduced to the system by using a hybrid semi-grand canonical Monte Carlo/Molecular Dynamics approach. This hybrid approach requires some properties to be calculated.

➤ Properties calculated with the new EAM potential:

- Melting Temperature of Ni: $T_m = 1760\text{ K}$ (compared to 1728 K)¹
- Equilibrium chemical potential difference: $\Delta\mu = \sim -0.55\text{ eV}$ of a supercooled system. (965 K)¹



¹ Hu et al. (2018) Nature Comm.

Summary and Conclusions

- A new embedded-atom method (EAM) interatomic potential parametrized for segregation induced embrittlement behavior of S in Ni is developed
 - The resulting fit is compared to literature with satisfactory results achieved
 - Allows simulation of a larger number of atoms or longer simulation times than DFT methods or atomistic simulation with the ReaxFF potential
- Segregation energies and embrittling potencies are calculated for six [100] STGBs using molecular statics calculations with the newly developed EAM potential
 - These results indicate the importance of the GB structure and S coverage on the embrittlement behavior
- Selection of GB sets by matching the embrittling potencies of different interfaces to isolate the role of grain boundary structure could provide insights on the embrittlement mechanisms¹