



# Top-Down Etch Processes for III-Nitride Nanophotonics

George T. Wang<sup>1</sup>, Benjamin Leung<sup>1</sup>, Miao-Chan Tsai<sup>2</sup>, Keshab R. Sapkota<sup>1</sup>, Barbara A. Kazanowska<sup>3</sup>, Kevin S. Jones<sup>3</sup>,

<sup>1</sup>Sandia National Laboratories, Albuquerque, NM, USA

<sup>2</sup>University of New Mexico, Albuquerque, NM

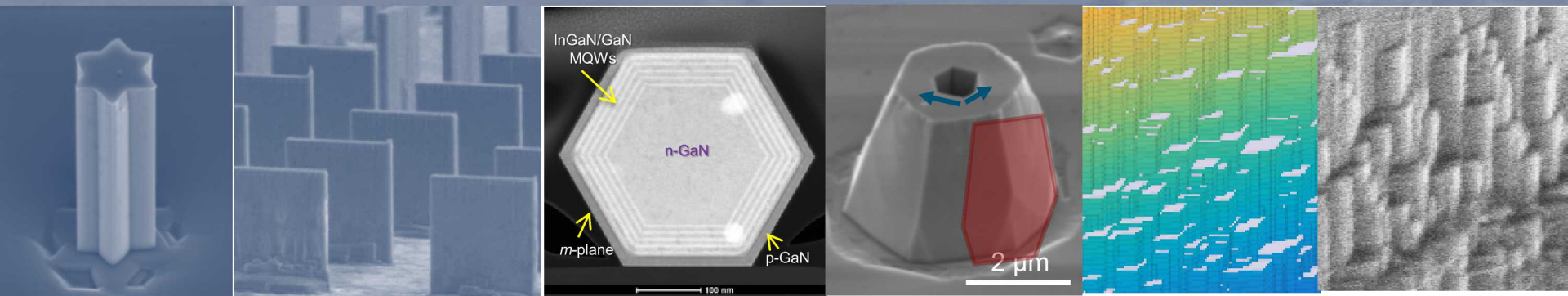
<sup>3</sup>University of Florida, Gainesville, FL

This work was funded by DOE BES Materials Science and Engineering Division, DOE Energy Frontier Research Center for Solid-State Lighting Science, and Sandia's LDRD program. This work was performed, in part, at the Center for Integrated Nanotechnologies, a U.S. Department of Energy, Office of Basic Energy

Sandia National Laboratories is a multimission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

# Outline

- Orientation-resolved etch rates
- Prediction of etch geometries
- Mechanism of highly anisotropic KOH wet etch
- Tapered phosphoric acid etch
- III-Nitride nanowire photonics



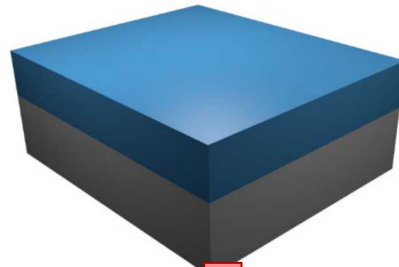
# Motivation: Top-down approach for nanostructures

**Motivation:** Advantages of *top-down approach* for fabrication, for both *material synthesis and integration*

## Steps

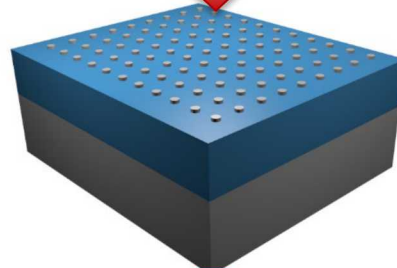
## Advantages

Conventional  
GaN film



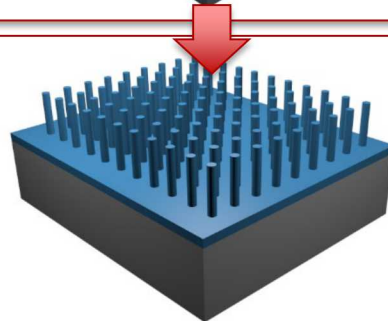
- Standard/optimized growth conditions (not limited by 1D growth window) give control over:
  - Dopant, impurity incorporation
  - Axial heterostructuring
- Can be characterized by standard methods

Lithographic  
technique



- Ordered and periodic arrays
- Independent control over height, width and pitch
- Cross-sectional shape control

Anisotropic  
etch

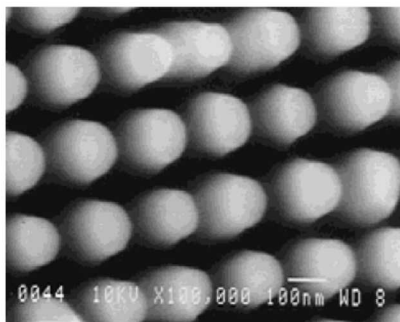


- Potentially **less tapered, higher aspect-ratio structures** possible for highly anisotropic, crystallographically selective etches



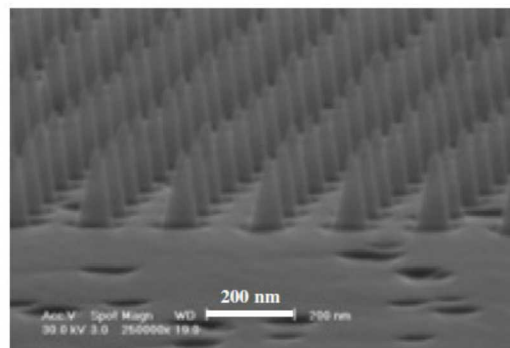
# Previous top-down nanopillars by dry etching

## InGaN/GaN MQW nanoposts/nanopillars



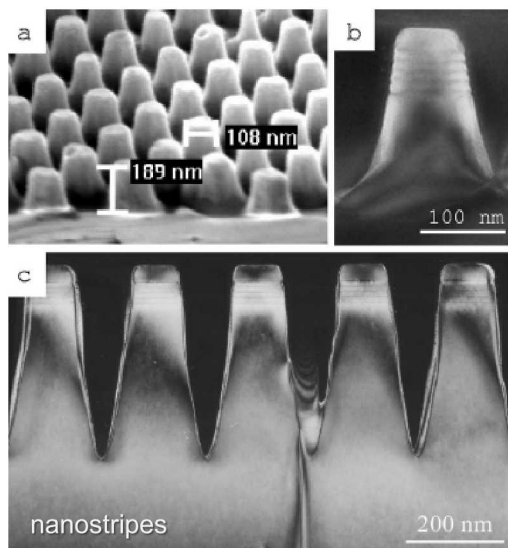
L. Chen et al., PSSa **188** 135 (2001) (Brown, Yale)

- E-beam Ni mask, Cl-based RIE etch



H. S. Chen et al., Nanotechnology **17** 1454 (2006) (NTU)

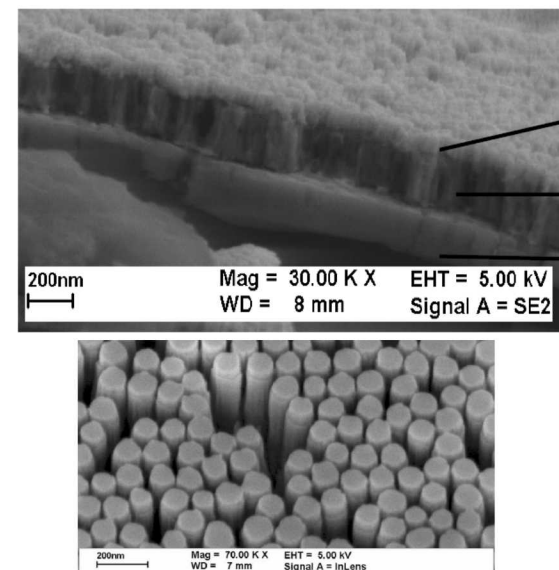
- E-beam Cr mask, Cl-based ICP-RIE etch
- Observed PL blue-shift w/decreasing diameter due to strain relaxation (reduction of QCSE)



S. Keller et al., J. Appl. Phys. **100** 054314 (2006) (USCB)

Holographic lithography, SiO<sub>2</sub> mask, CHF<sub>3</sub> ICP etch to make strips and pillars

## InGaN/GaN MQW nanorod LED



C. Y. Wang et al., Opt. Expr. **16**, 10549, 2008. (NTU)

M. Y. Ke et al., IEEE J. Sel. Top. Quantum Elec. **15** 1242 (2009) (NTU)

SiO<sub>2</sub> nanosphere lithography, Cl-based ICP-RIE etch; electrically injected array

- **Tapered, short aspect ratio nanopillars via plasma etching**
- **Plasma etch causes sidewall damage**



# Chemical wet etching of the III-Nitrides: Is it useful ?

Wet etching typically not considered viable due to very low etch rates ( $< \text{\AA}/\text{min}$ )

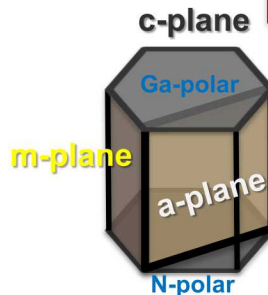
Significant etch rates, but extremely crystallographically anisotropic

Table 1  
GaN and AlN etching results in acid and base solutions

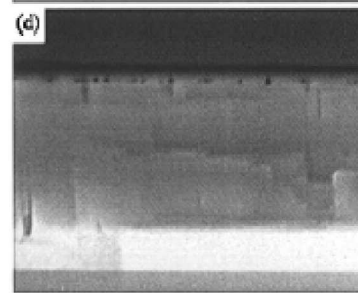
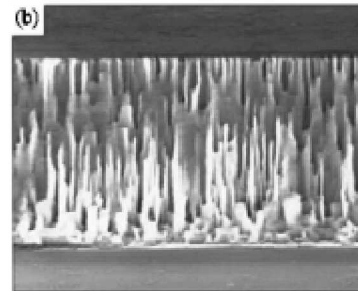
Etching solutions	GaN etch rate (nm/min)
Citric acid	0 (75 °C)
Succinic acid	0 (75 °C)
Oxalic acids	0 (75 °C)
Nitric acid	0 (85 °C)
Phosphoric acid	0 (82 °C)
Hydrochloric acid	0 (80 °C)
Hydrofluoric acid	0
Hydroiodic acid	0
Sulfuric acid	0 (82 °C)
Hydrogen peroxide	0
Potassium iodide	0
2% Bromine/methanol	0
<i>n</i> -Methyl-2-pyrrolidone	0
Sodium hydroxide	0
Potassium hydroxide	0
AZ400K photoresist developer	0
Hydroiodic acid/hydrogen peroxide	0
Hydrochloric acid/hydrogen peroxide	0
Potassium triphosphate	0 (75 °C)
Nitric acid/potassium triphosphate	0 (75 °C)
Hydrochloric acid/potassium triphosphate	0 (75 °C)
Boric acid	0 (75 °C)
Nitric/boric acid	0 (75 °C)
Nitric/boric/hydrogen peroxide	0
HCl/H <sub>2</sub> O <sub>2</sub> /HNO <sub>3</sub>	0
Potassium tetra borate	0 (75 °C)
Sodium tetra borate	0 (75 °C)
Sodium tetra borate/hydrogen peroxide	0
Potassium triphosphate	0 (75 °C)
Potassium triphosphate/hydrogen peroxide	0

Etching was conducted at room temperature (25 °C) unless otherwise noted (after [31]).

But actually...

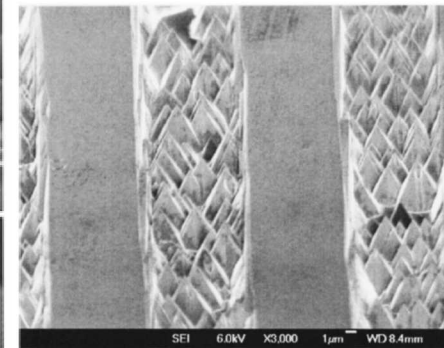


m-plane sidewalls



Stocker et al. APL 77, 4253 (2000)

Ga-polar and N-polar GaN



H. Ng et al., J. Appl. Phys., 94, 650 (2003)

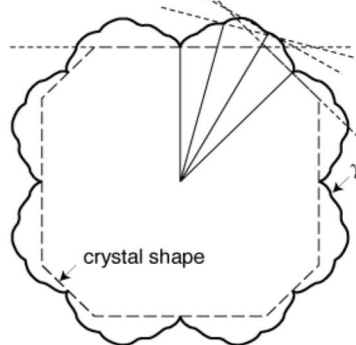
KOH-based wet etches

Anisotropy is **> 2000:1** (GaN/KOH), as compared to maximum anisotropies of **160:1** (Si), and **4:1** (GaAs)

This property, however, makes wet etching *ideal for micro/nanostructure fabrication of the III-nitrides* !

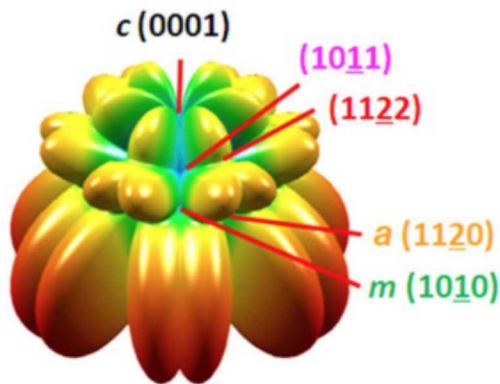
# The appearance and evolution of facets

Predicting the equilibrium crystal shape: **The Wulff Plot**



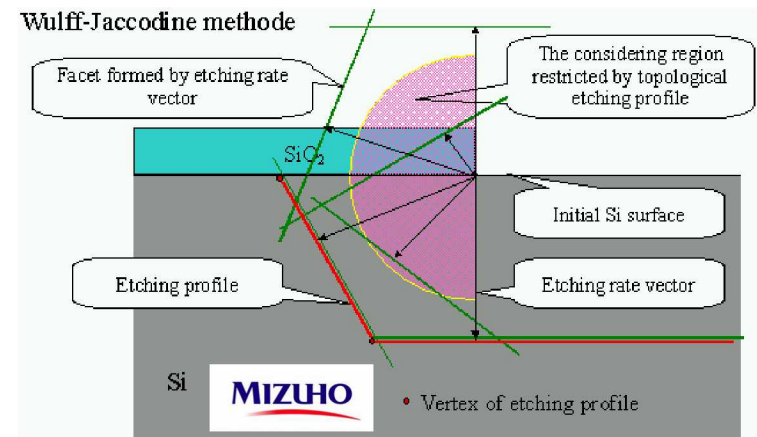
Wulff et al., Semicond. Z. Kristallogr 34, 449 (1901)

For MOVPE growth, use the *kinetic* Wulff plot



B. Leung et al., Semicond. Sci. Tech. 27, 024005 (2012)

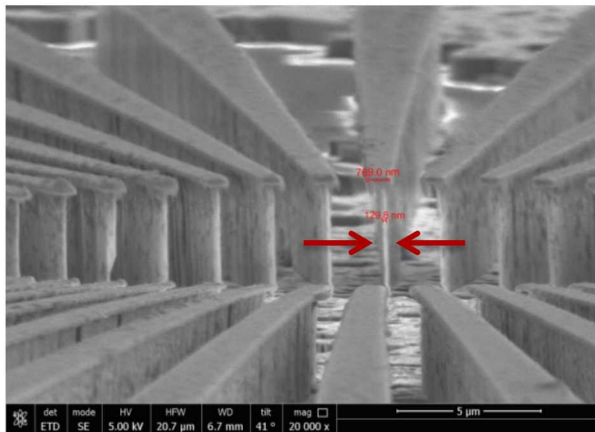
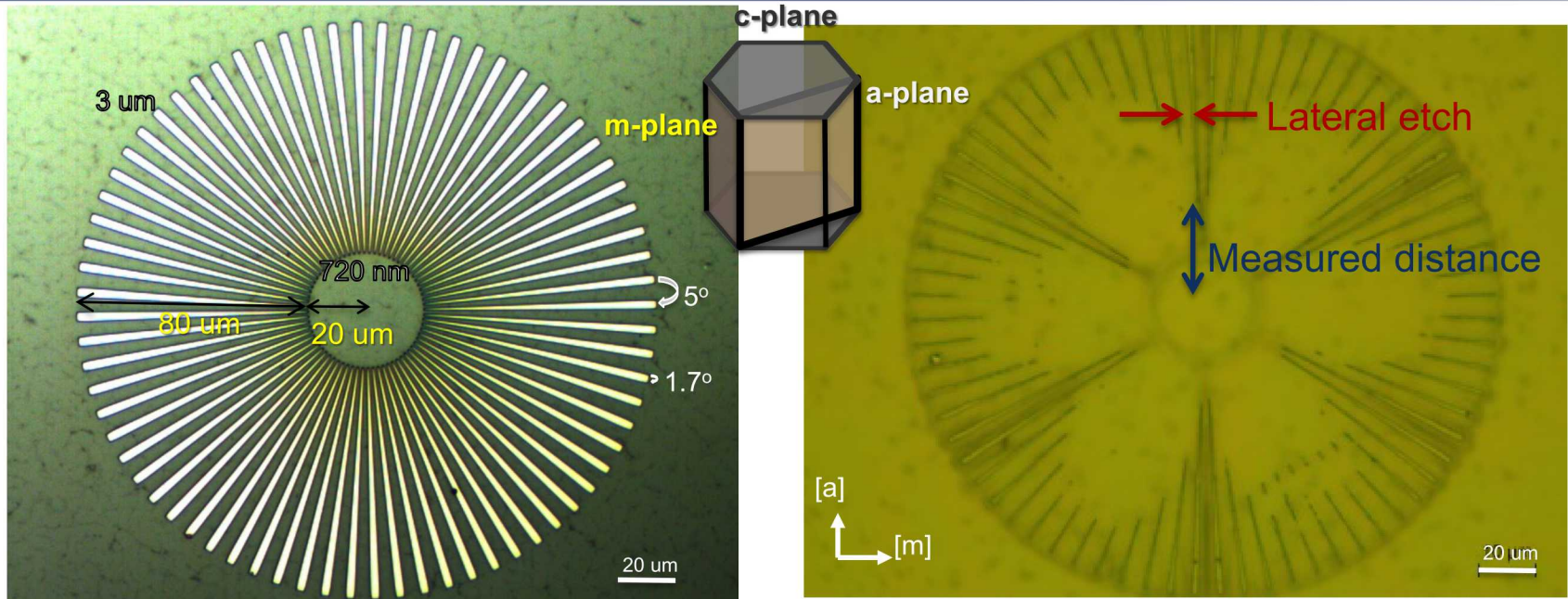
Same method used in commercial silicon wet etch simulators



Can we map the “Wulff plot” for etching?



# Directly Measuring Orientation Resolved Etch Rates



1.7°  
80  $\mu\text{m}$   
3  $\mu\text{m}$   
Lateral etch of 100 nm/min  $\rightarrow$   
Wedge retraction of 6.7  $\mu\text{m}/\text{min}$

5% KOH in ethylene glycol

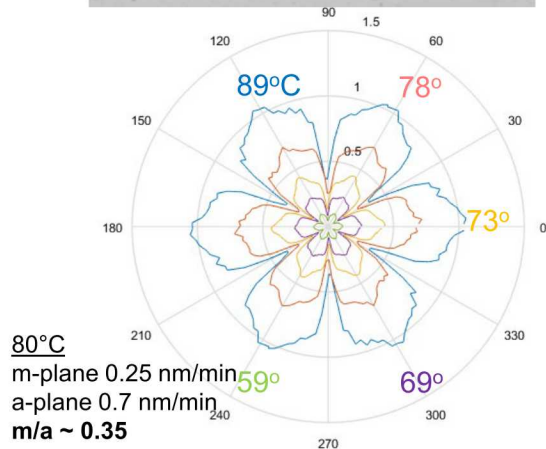
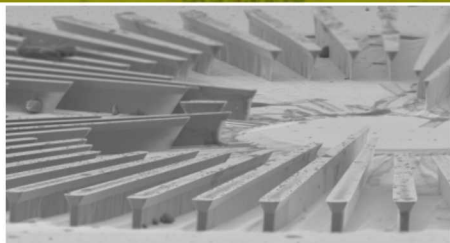
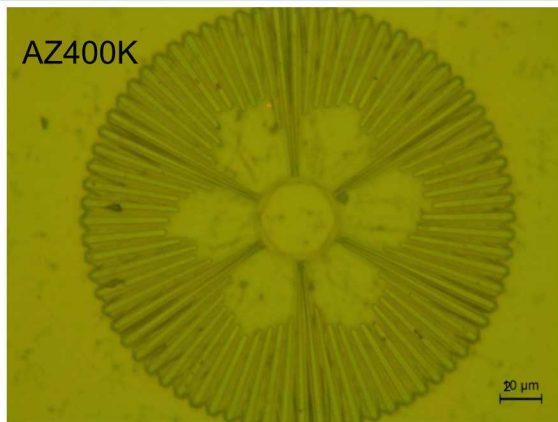
Etch rate is amplified  
by a factor of 67

Direct measurement of orientation resolved etch  
rates for the first time in GaN

Wagon wheel methodology: see for example: Wind, R. A.; Hines, M. A., *Surf. Sci.* 2000, 460, 21.



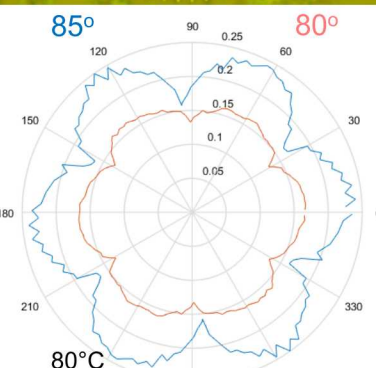
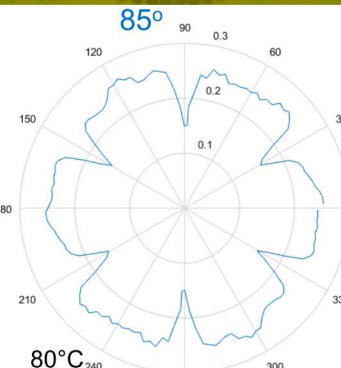
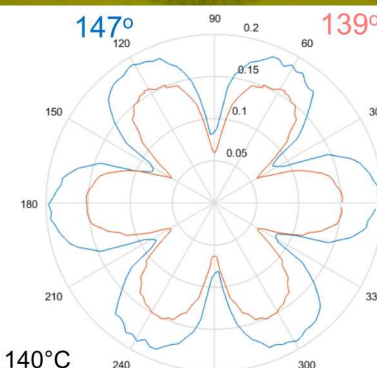
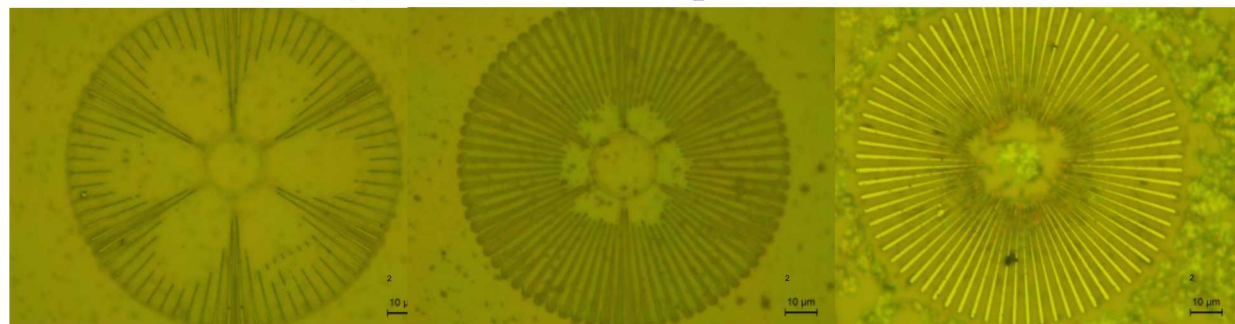
# Direct Measurements of Orientation Resolved Etch Rates – Chemistries and Temperatures



5% KOH in ethylene glycol

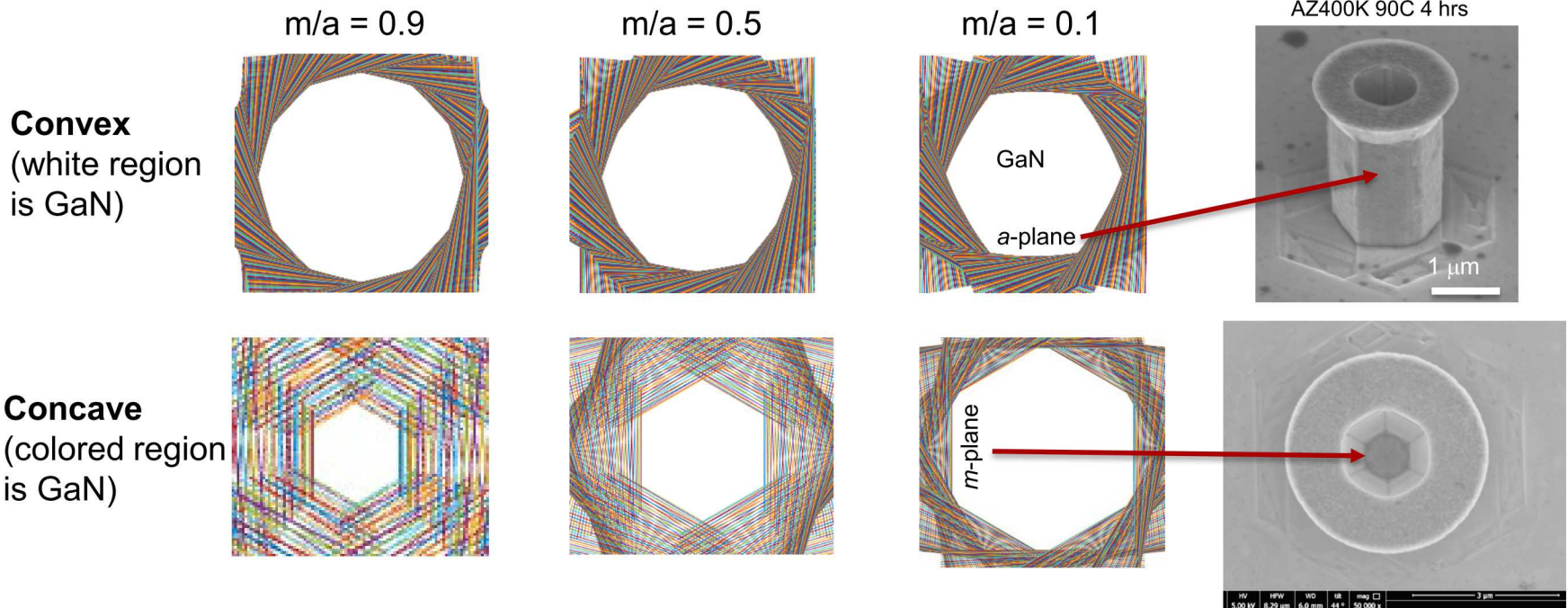
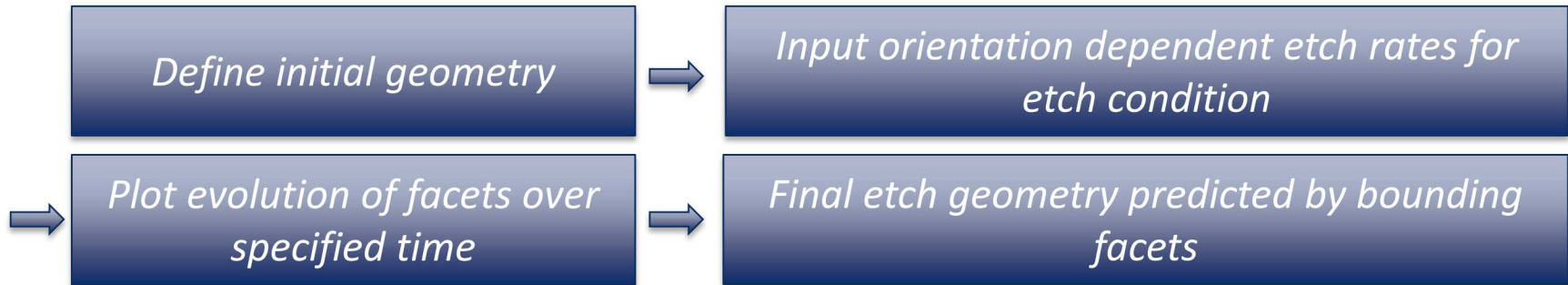
5% KOH in H<sub>2</sub>O

25% TMAH



*m-plane*: Atomically smooth etch facet  
*a-plane*: Transition from step-flow highly  
 dependent on chemistry  
*m/a*: 0.3 ~ 0.9

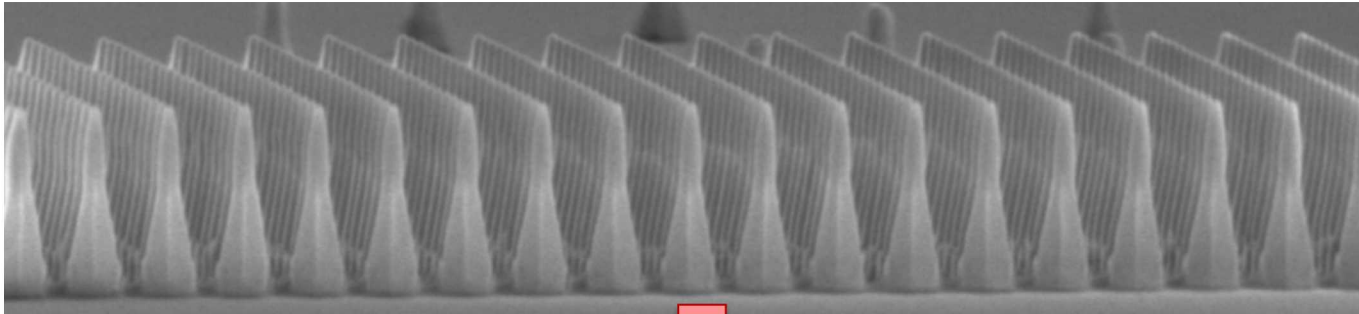
# Prediction of Etched Geometry: Implementing the Wulff-Jaccodine Method



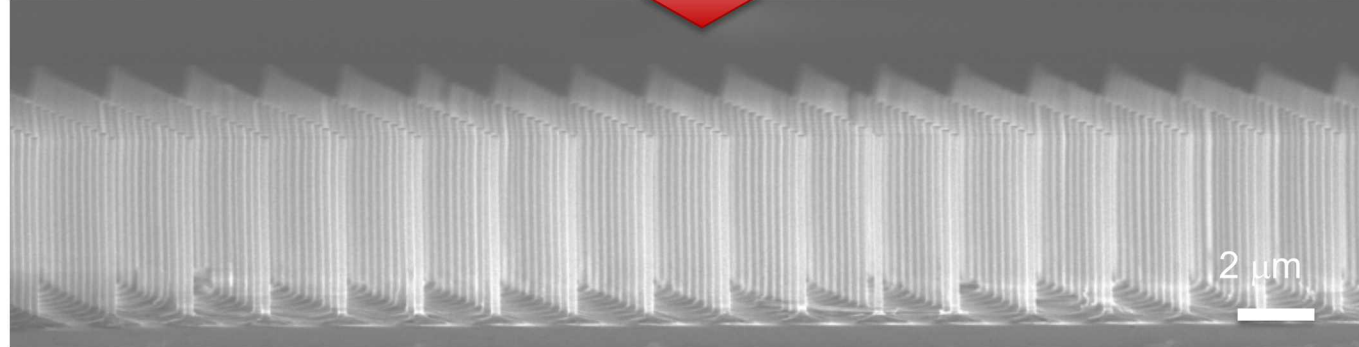


# Fabricated top-down GaN nanostructures using two-step dry + wet etch approach

Dry etch ( $\text{Cl}_2$  based ICP)



KOH-based wet etch (AZ400K)



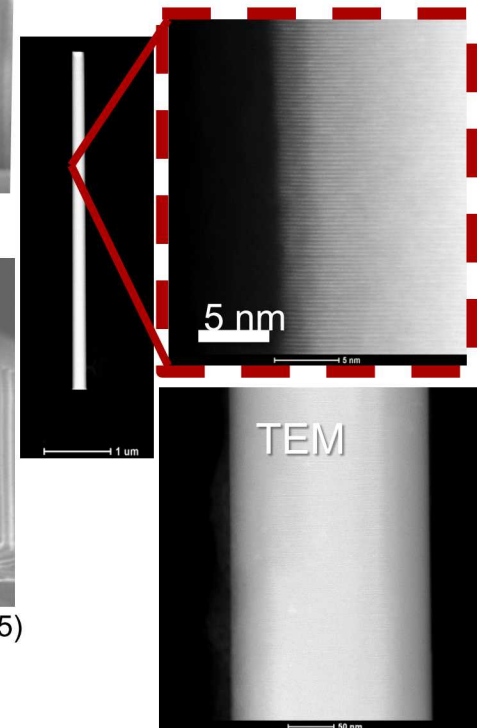
Diameter: 180 nm, Height: 4.5 μm, Pitch: 2 μm (AR: 25)

See: Q. Li et al., Opt. Exp., 20, 17873 (2012)

Q. Li et al., Opt. Exp., 19, 25528 (2011)

etc....

TEM shows smooth vertical sidewalls



What is the wet-etch mechanism from *rough tapered* to *smooth vertical sidewalls*?



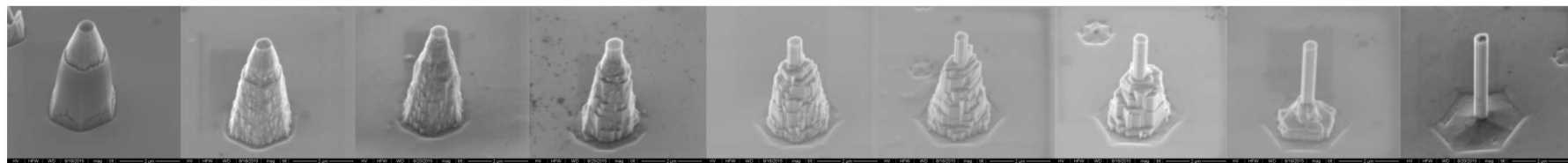
# Wet etch evolution for nanowires

As ICP etched    1 min    2 min    5 min    10 min    15 min    20 min    30 min    52 min

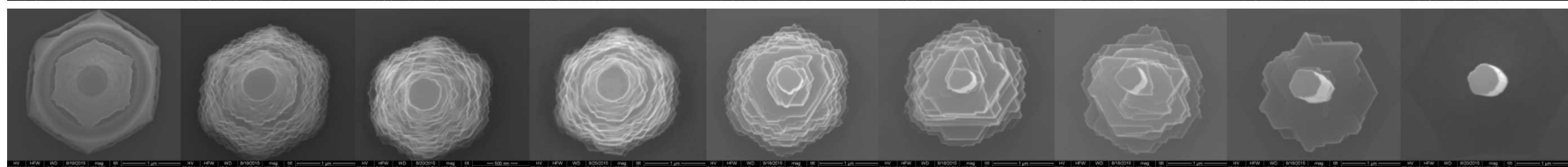
Side  
view



45°



Top  
view



AZ400K, 65°C

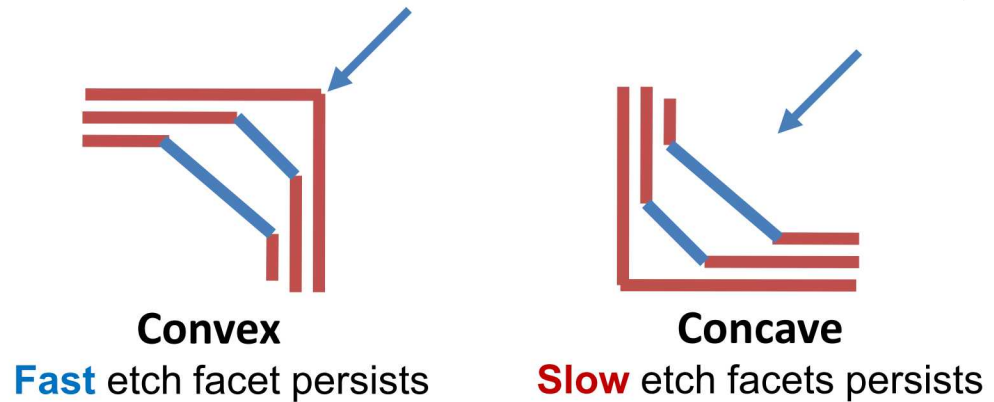
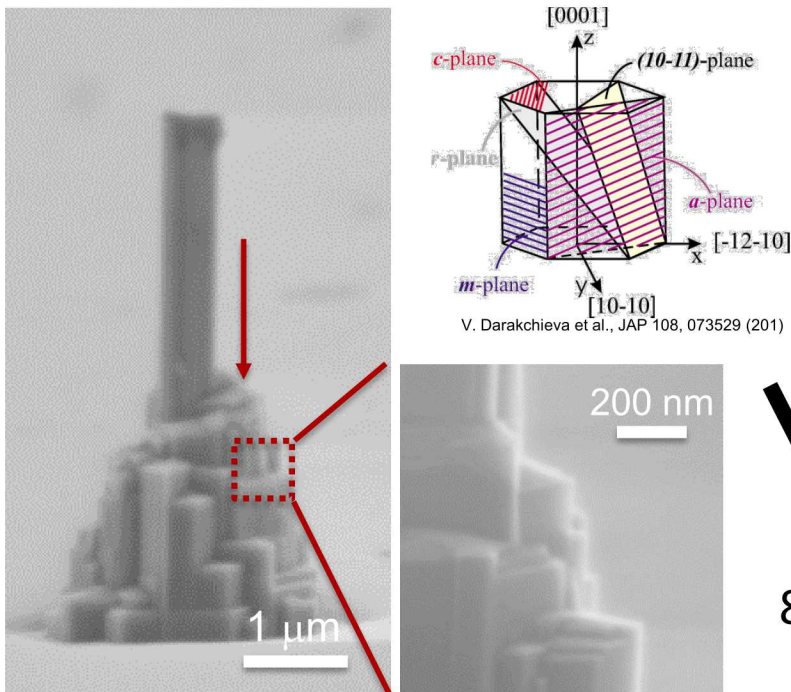
Wet etch proceeds “vertically” rather than horizontally

# Mechanism/Basis for evolution of facets during GaN wet etching

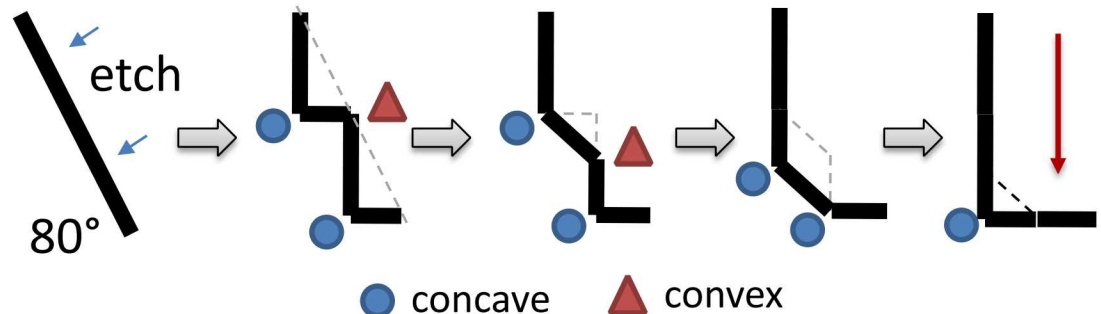
What is the mechanism from *tapered to smooth vertical sidewalls* through faceting?

*Slow etching: c-plane, m-plane*

*Fast etching: semipolar (10 $\bar{1}1$ )*



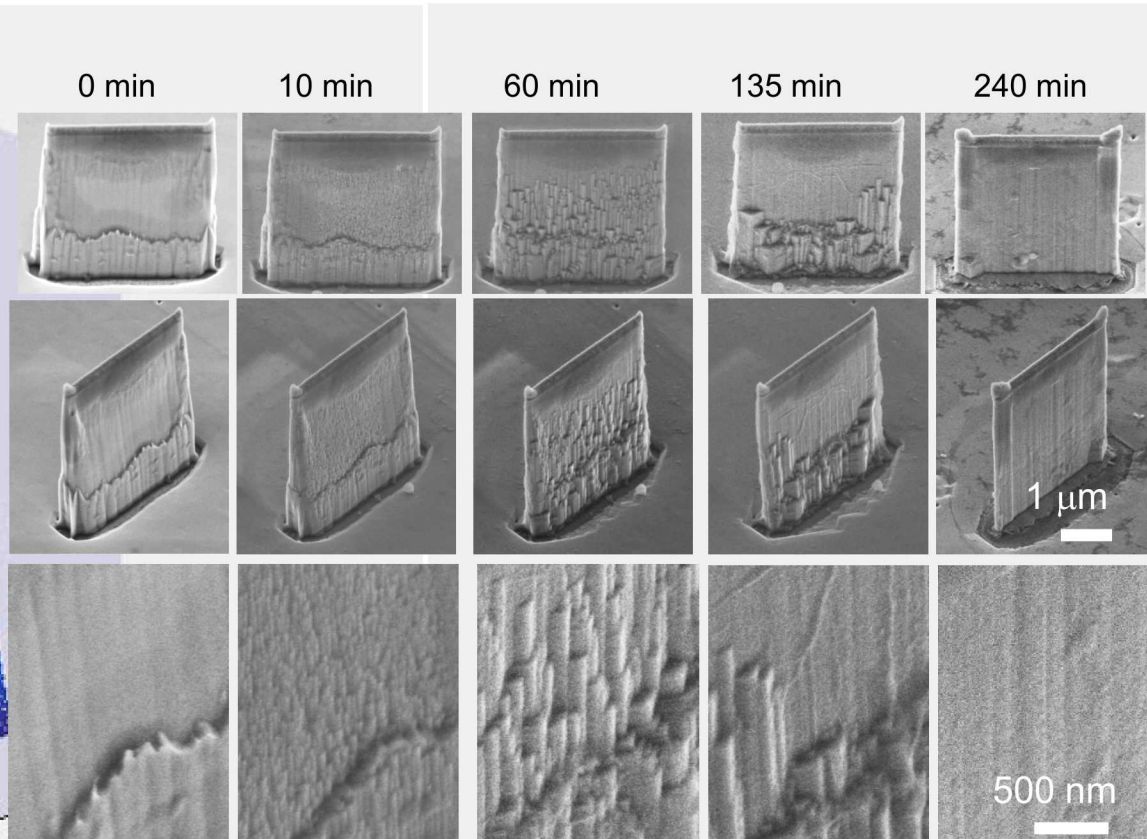
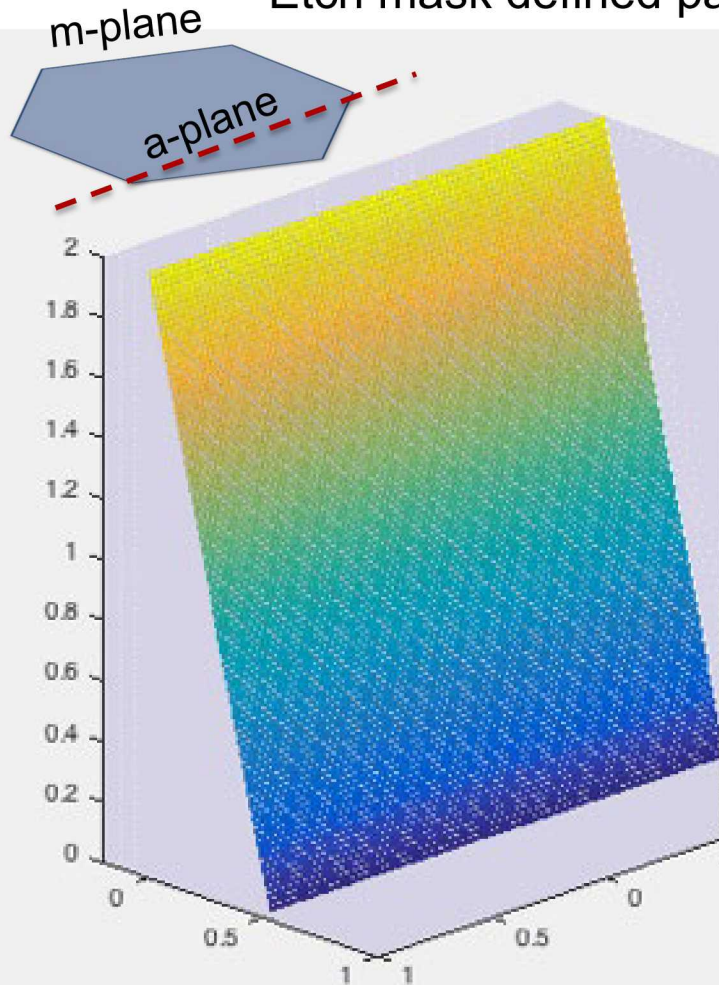
**The facet etch sequence:**



The appearance (disappearance) of fast (slow) etching facets in concave (convex) geometries is the basis of the etch mechanism

# Etch evolution of $a$ -plane wall

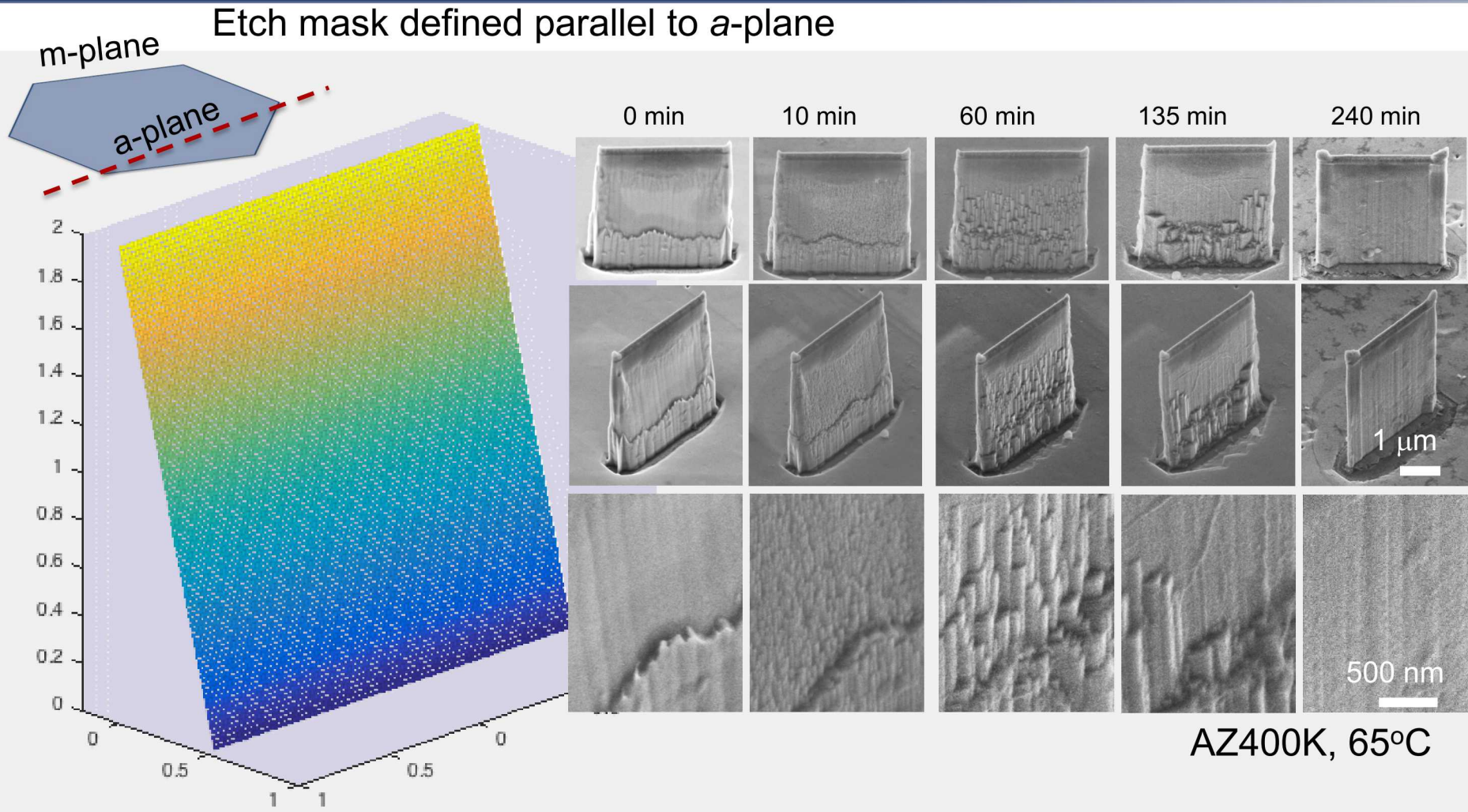
Etch mask defined parallel to  $a$ -plane



AZ400K, 65°C

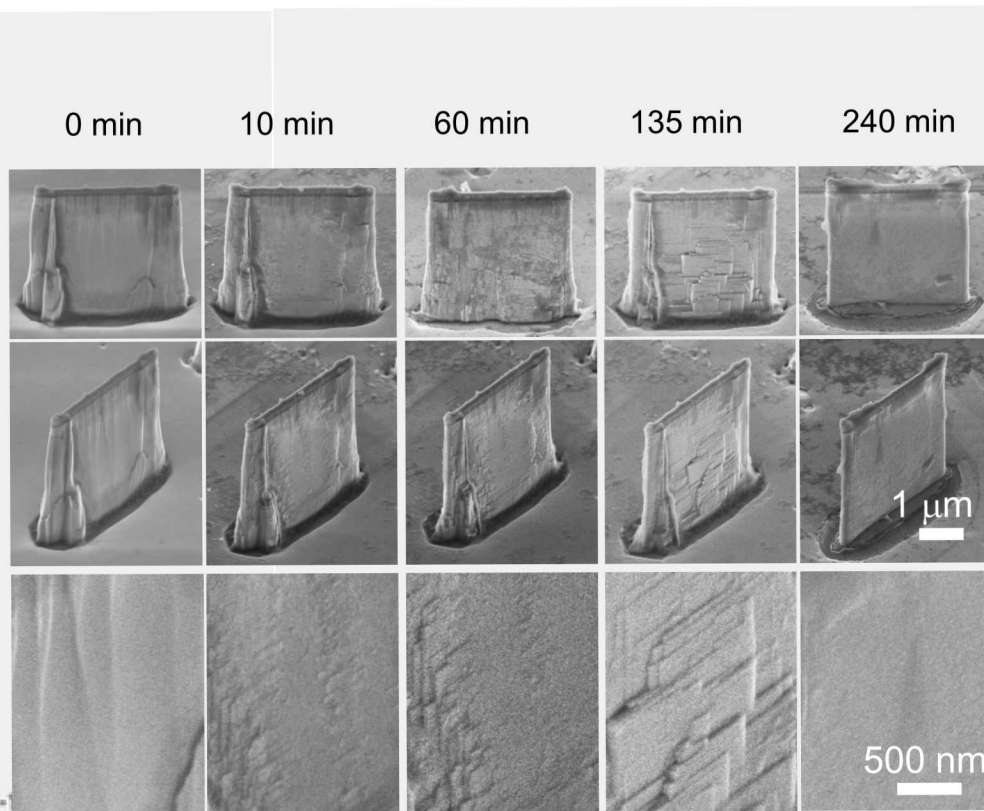
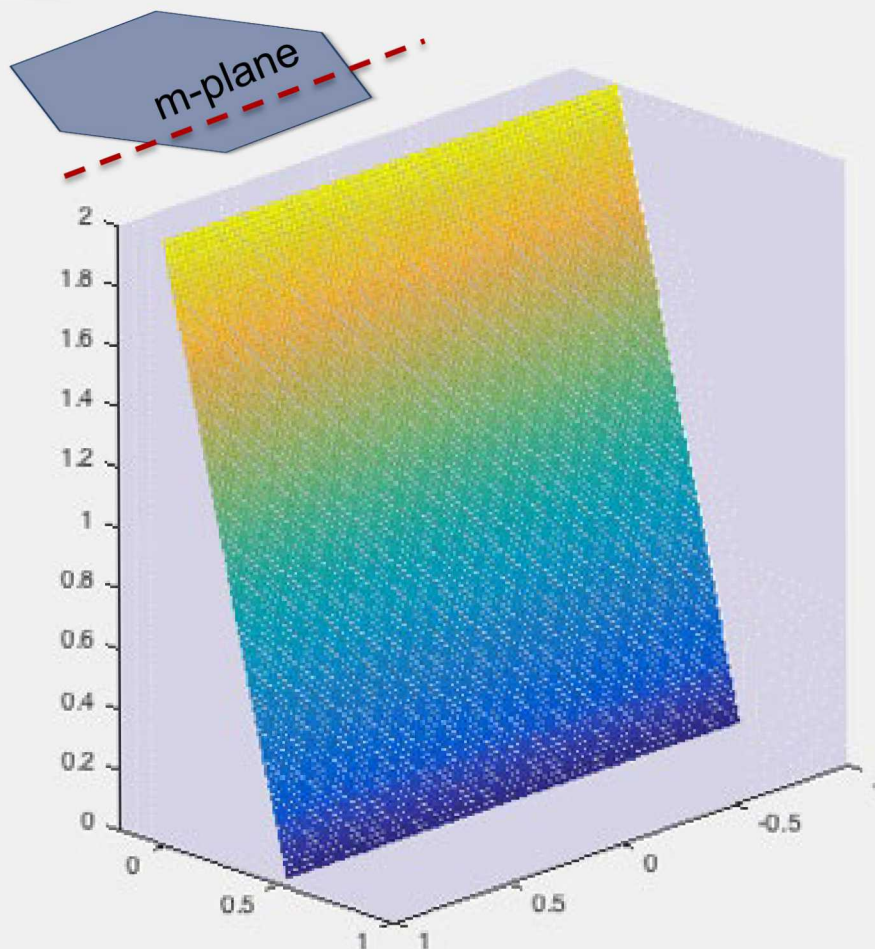


# Etch evolution of $a$ -plane wall



# Etch evolution of $m$ -plane wall

Etch mask defined parallel to  $m$ -plane

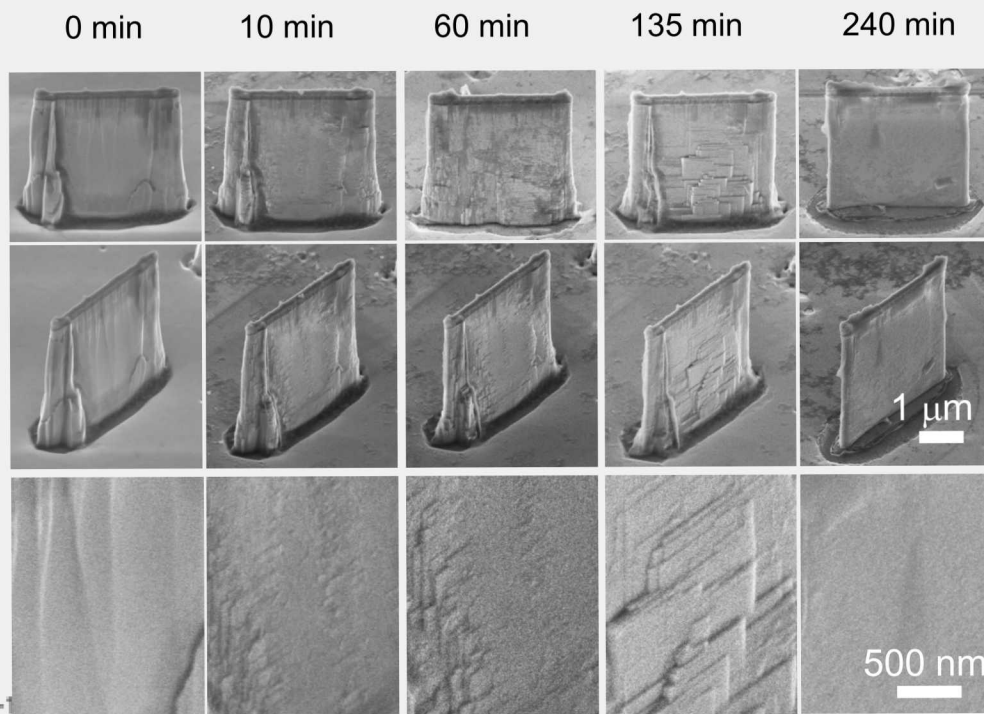
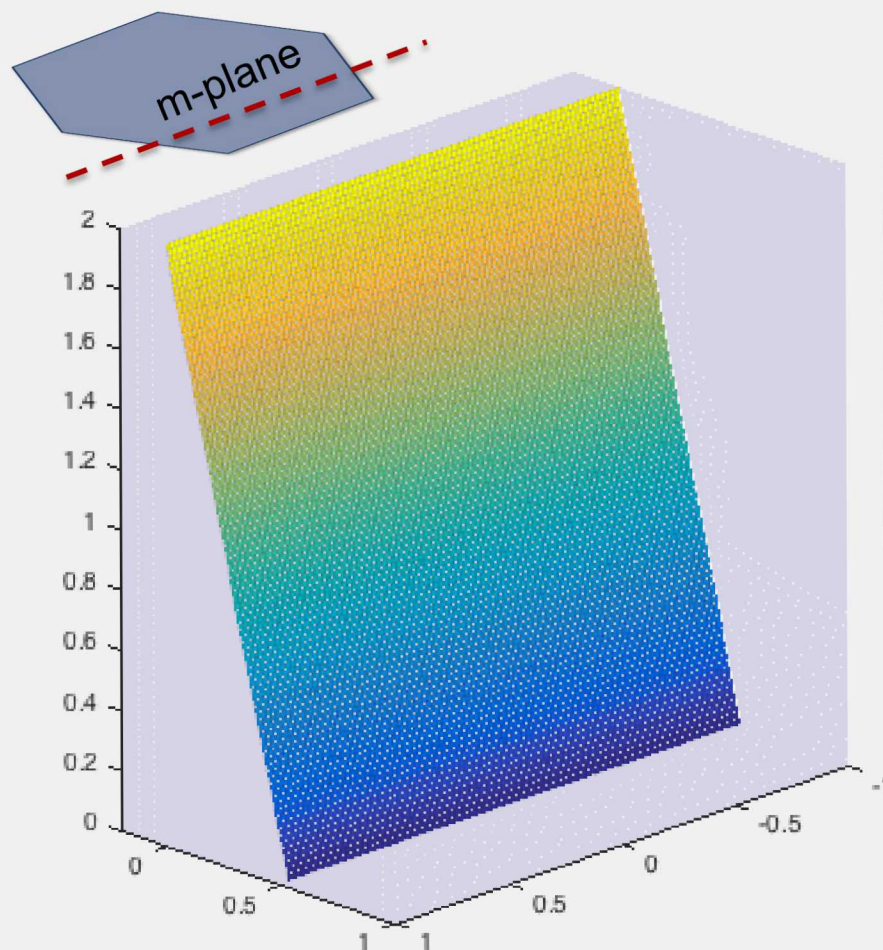


AZ400K, 65°C



# Etch evolution of $m$ -plane wall

Etch mask defined parallel to  $m$ -plane

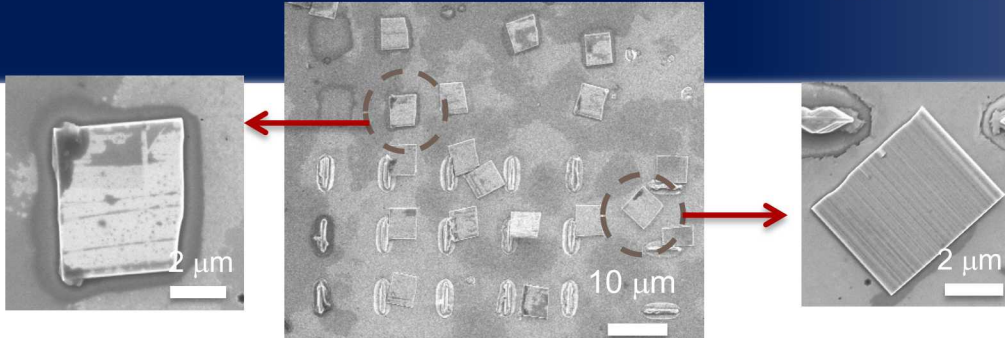


AZ400K,  $65^\circ\text{C}$



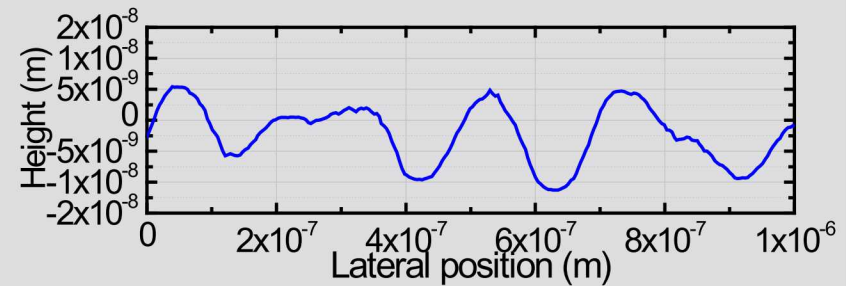
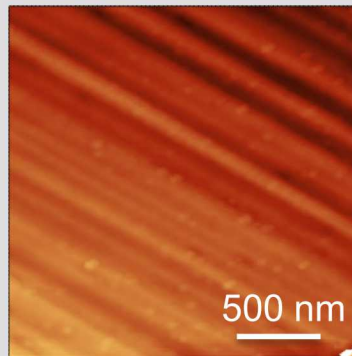
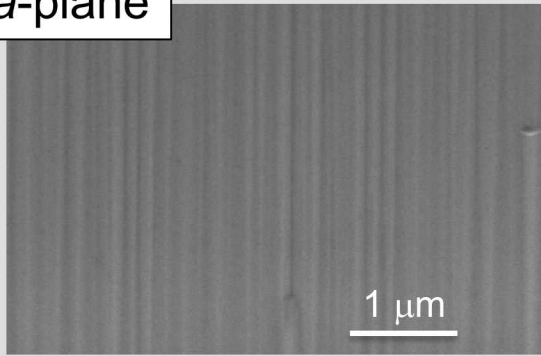
# Measured etched nanowall morphology by AFM

*m*-plane wall



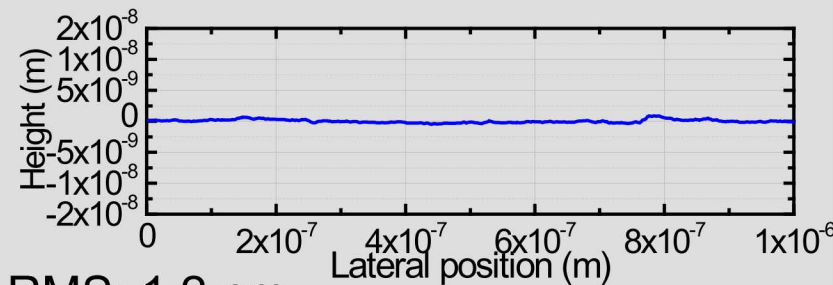
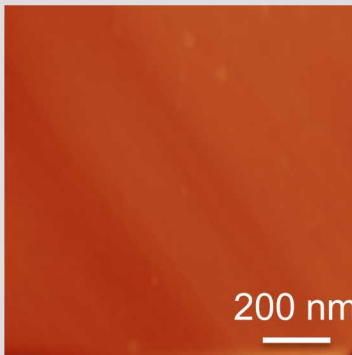
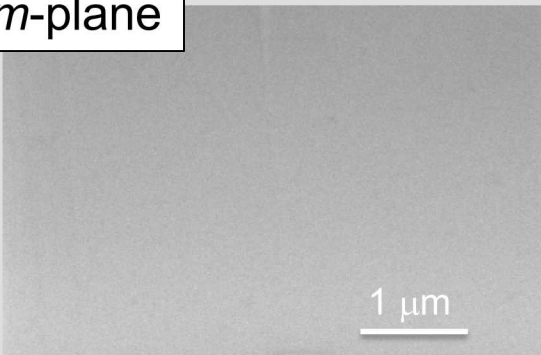
*a*-plane wall

*a*-plane



RMS: 7.6 nm

*m*-plane

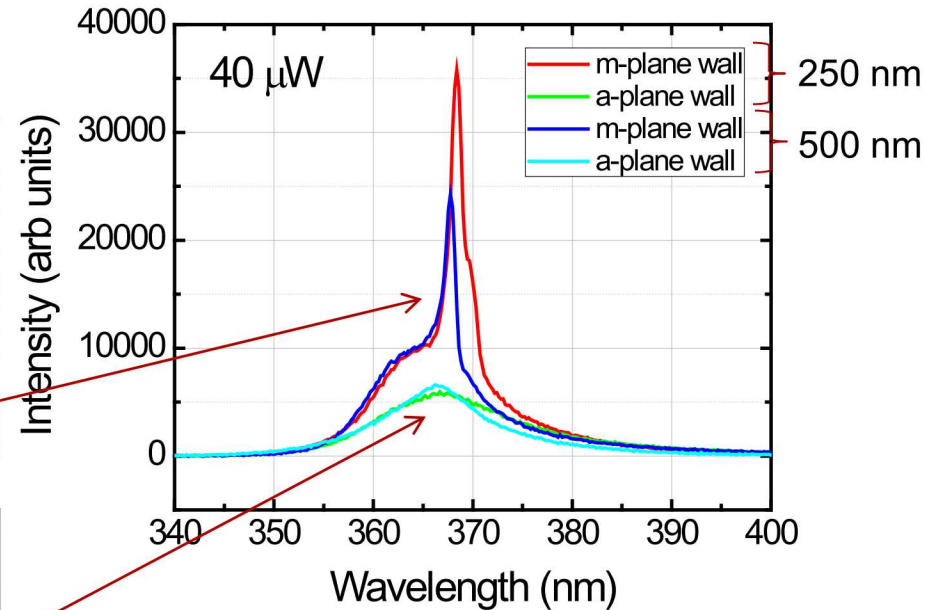
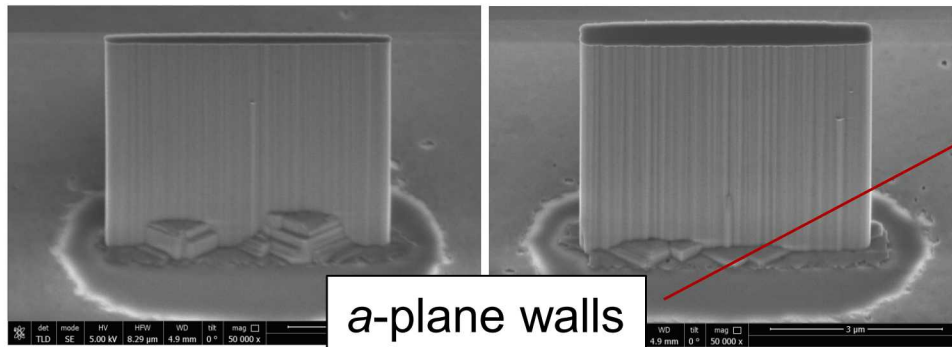
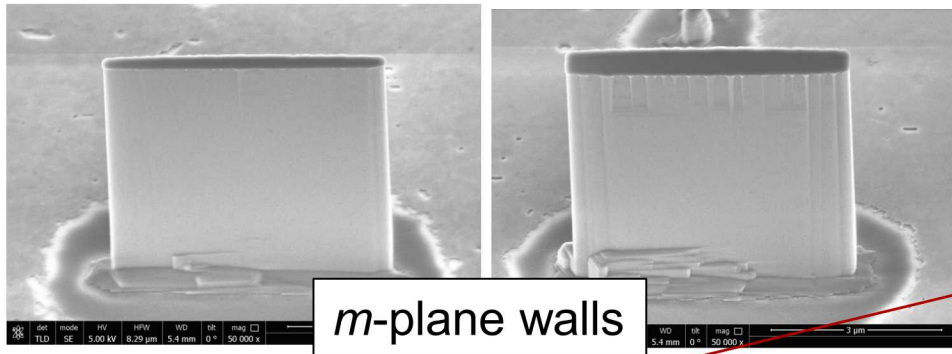


RMS: 1.2 nm

Achieved atomically smooth surfaces through a two-step top-down process!

# Cavity modes in nanowalls

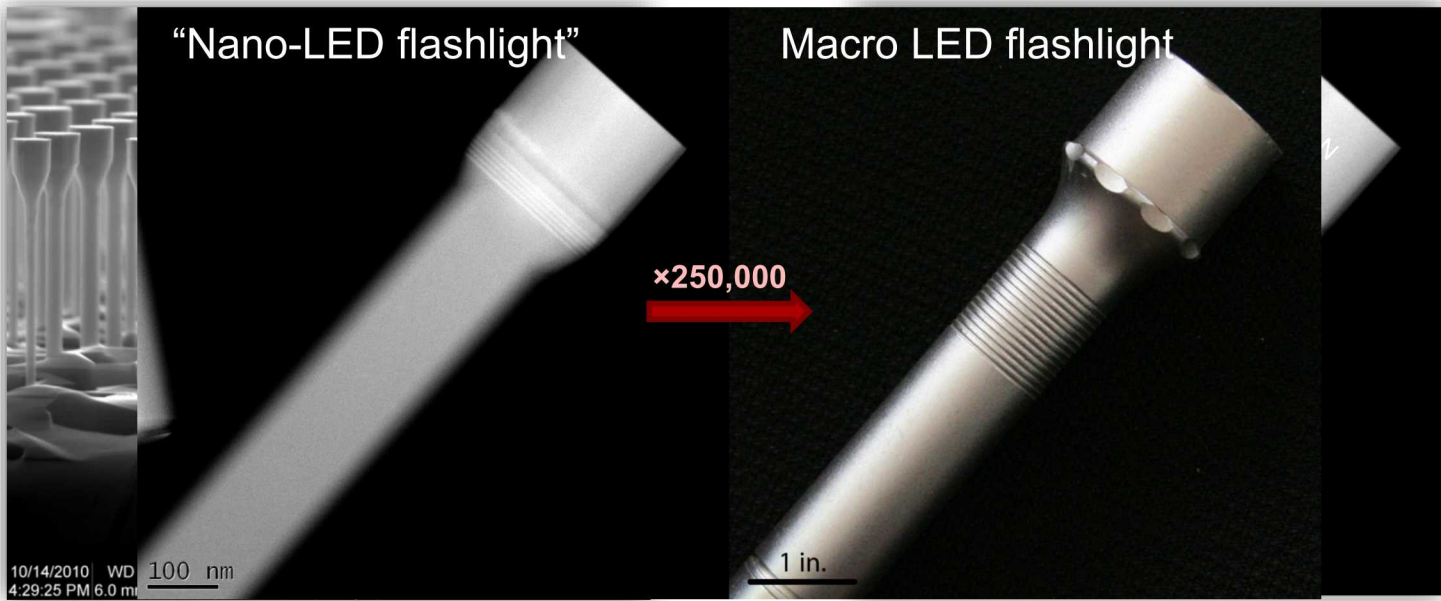
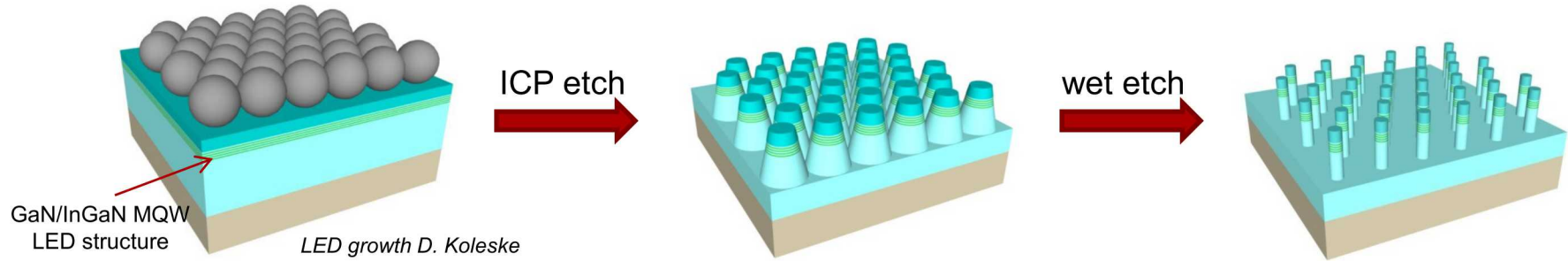
Optical pumping of GaN nanowalls  
Nd:YAG, 266 nm pulsed laser,  $\mu$ PL



Optically pumped nanowalls shows luminescence coupled to cavity modes only with the smooth m-plane sidewalls

Etched surface morphology has important effects on optical properties of nanostructures

# Axial GaN/InGaN nanowire LEDs



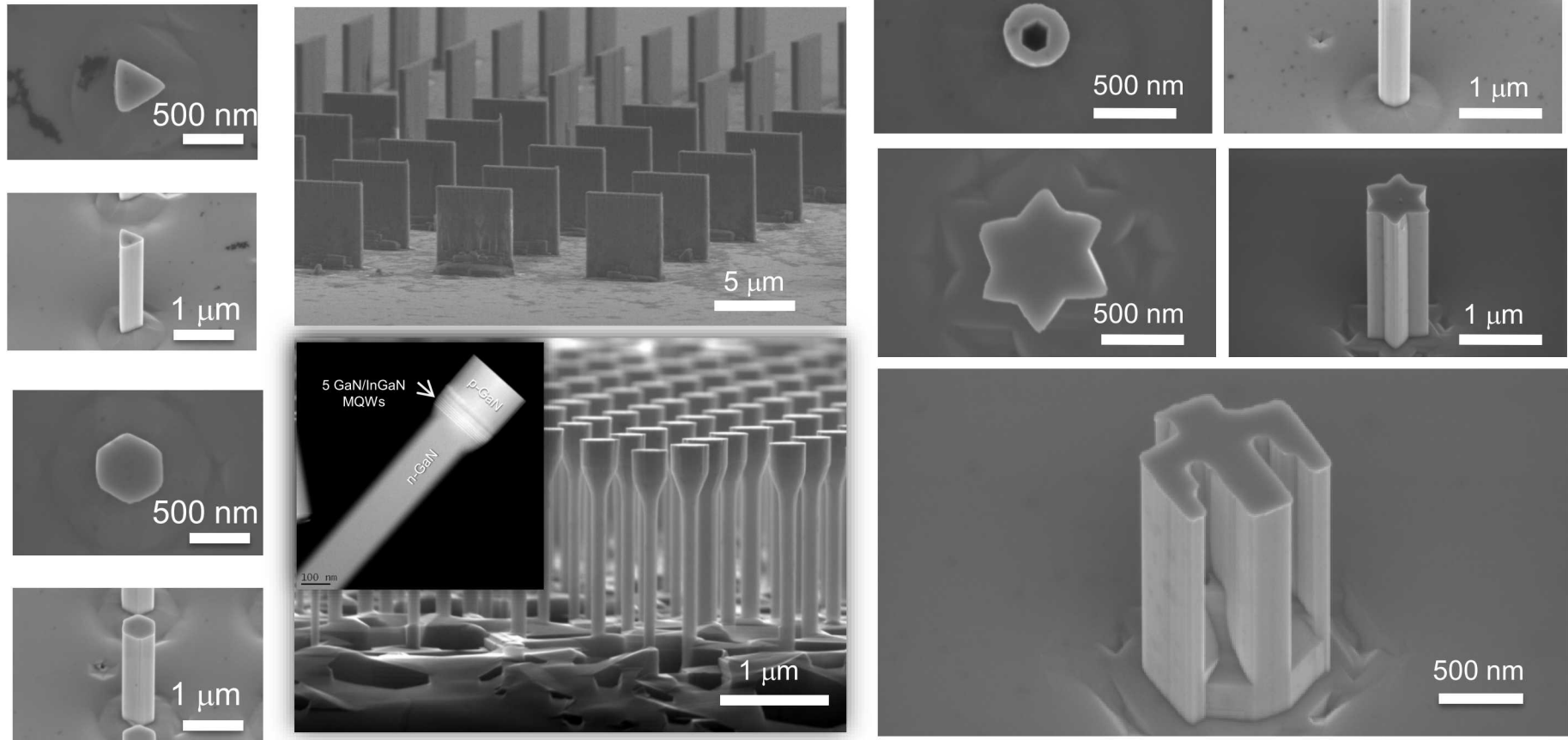
Q. Li et al.,  
*Optics Express* **19**,  
25528 (2011)

Doping affects etch rate – provides opportunity for 3D shape engineering



# Controlled and complex cross-sections

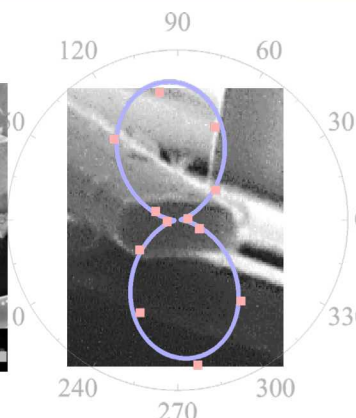
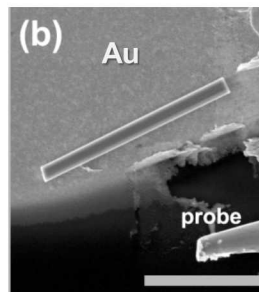
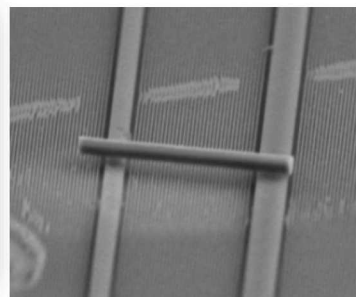
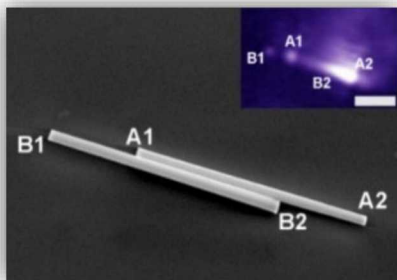
Can we expand to more *complex geometries* ?



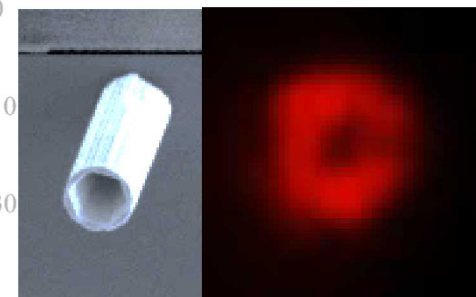
Understanding of the etch evolution enables design of a wide range of smooth faceted vertical nanostructures

# Top down vertical nanostructures for photonics

## Mode & polarization control in GaN nanowire lasers



## Beam shape control nanotube lasers



Q. Li et al., *Optics Express* **20** 17874 (2012)

H. Xu et al., *Appl. Phys. Lett.* **101** 221114 (2012)

H. Xu et al., *Appl. Phys. Lett.* **101** 113106 (2012)

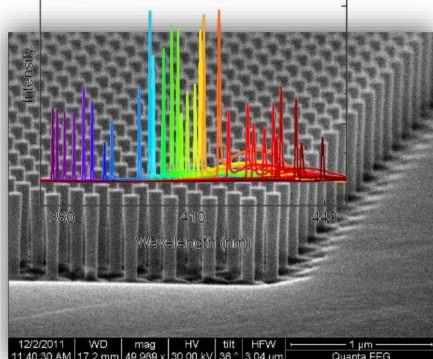
H. Xu et al., *Opt. Exp.* **22**, 19198 (2014)

J.B. Wright et al., *Appl. Phys. Lett.*, 104, 041107 (2014)

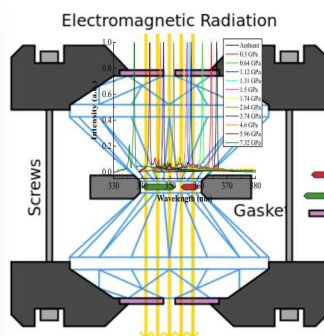
Li, Changyi, et al.  
*Nanoscale* **8**, 5682 (2016)

C. Li et al., *ACS Photonics*, **2**, 1025 (2015).

## Tunable wavelength nanowire lasers

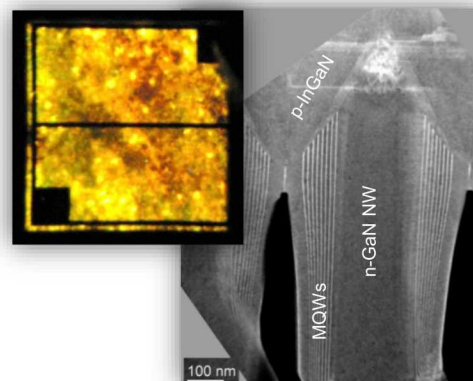


J.B. Wright et al., *Sci. Reports* **3**, Art no. 2982 (2013) doi:10.1038/srep02982



S. Liu, C. Li, J. J. Figiel, S. R. Brueck, I. Brener, G. T. Wang, *Nanoscale*, **7**, 9581 (2015).

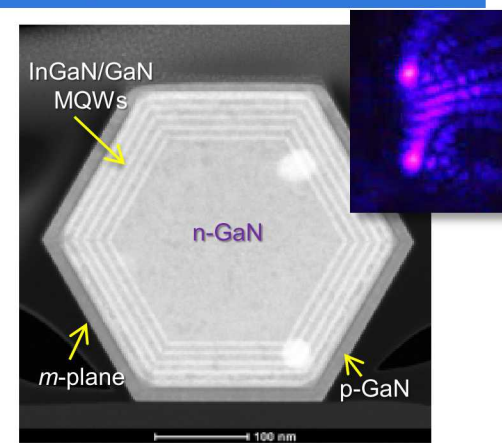
## Nonpolar core-shell nanowire LEDs, solar cells, and lasers



J. Wierer et al., *Nanotechnology* **23** 194007 (2012)

Riley, J.; *Nano Lett.* **14**, 4317 (2013).

G. T. Wang et al., *Phys. Stat. Solidi A*, **211**, 748 (2014)



C. Li et al., *Nano Lett.* **17**, 1049 (2017)

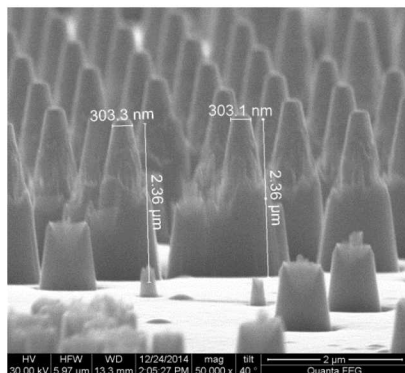
Howell, S. L., *Nano Lett.*, **13**, 5123 (2013)

# Top-down fabrication of AlGaN nanowires?

After ICP etch only

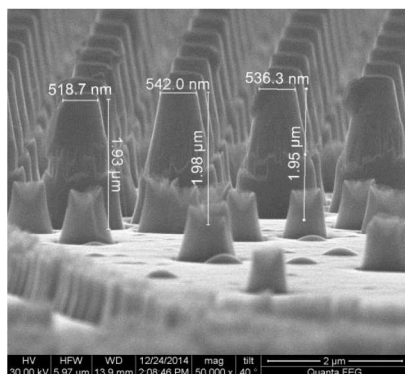
**30% AlGaN**

4.0  $\mu\text{m}$   $\text{Al}_{0.31}\text{Ga}_{0.69}\text{N}$   
on 1.6  $\mu\text{m}$  AlN on c-  
sapp



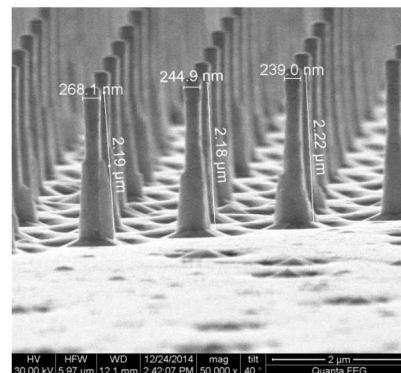
**70% AlGaN**

3.0  $\mu\text{m}$   $\text{Al}_{0.71}\text{Ga}_{0.29}\text{N}$   
on 1.5  $\mu\text{m}$  AlN on c-  
sapp

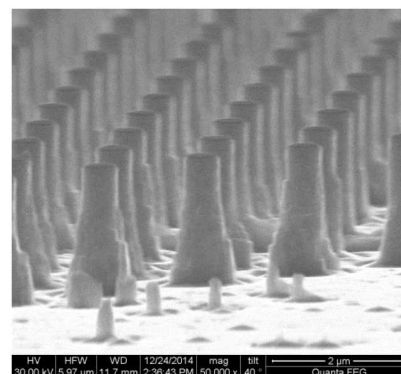
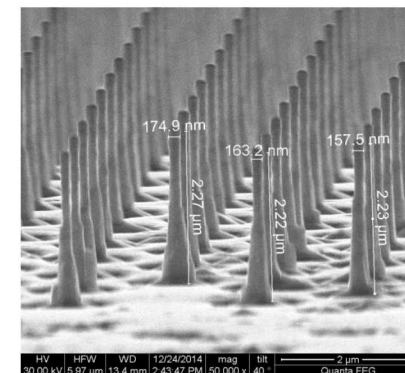


After AZ400K 65°C wet etch

240-500nm diameter



150-250 nm diameter

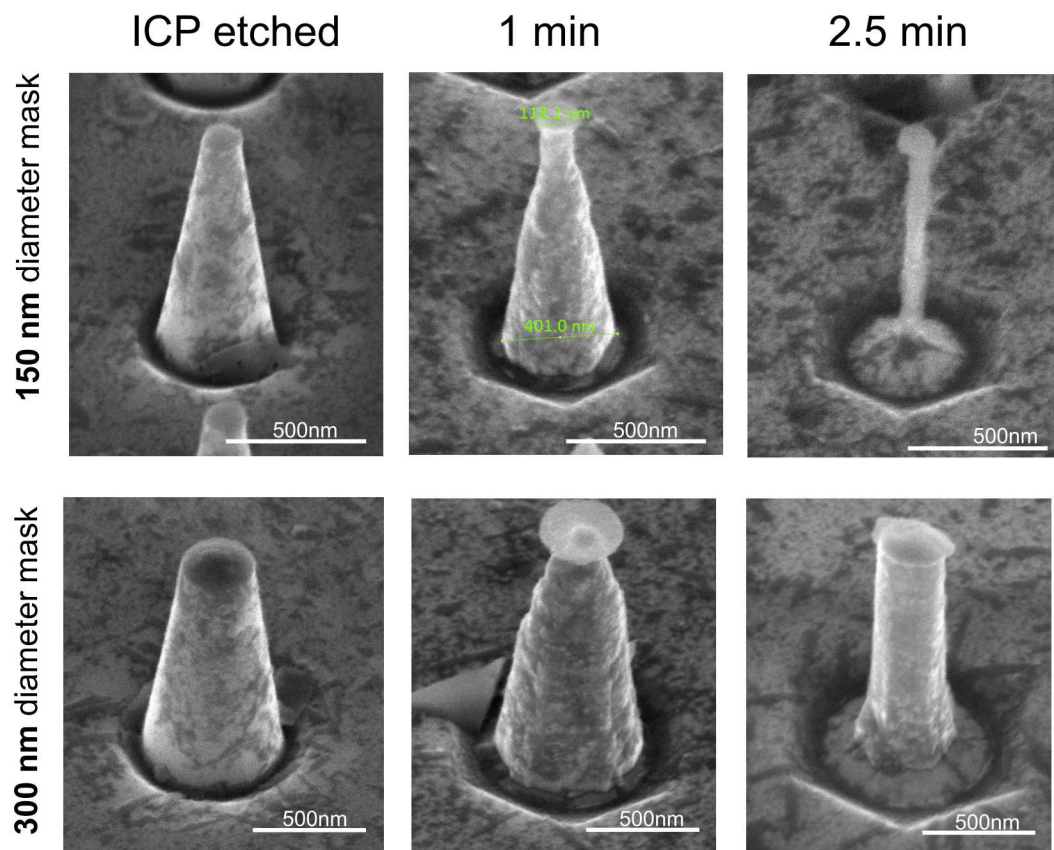


AlGaN samples grown by Andrew Allerman

**AlGaN exhibits rougher, less vertical sidewalls than GaN – why??**



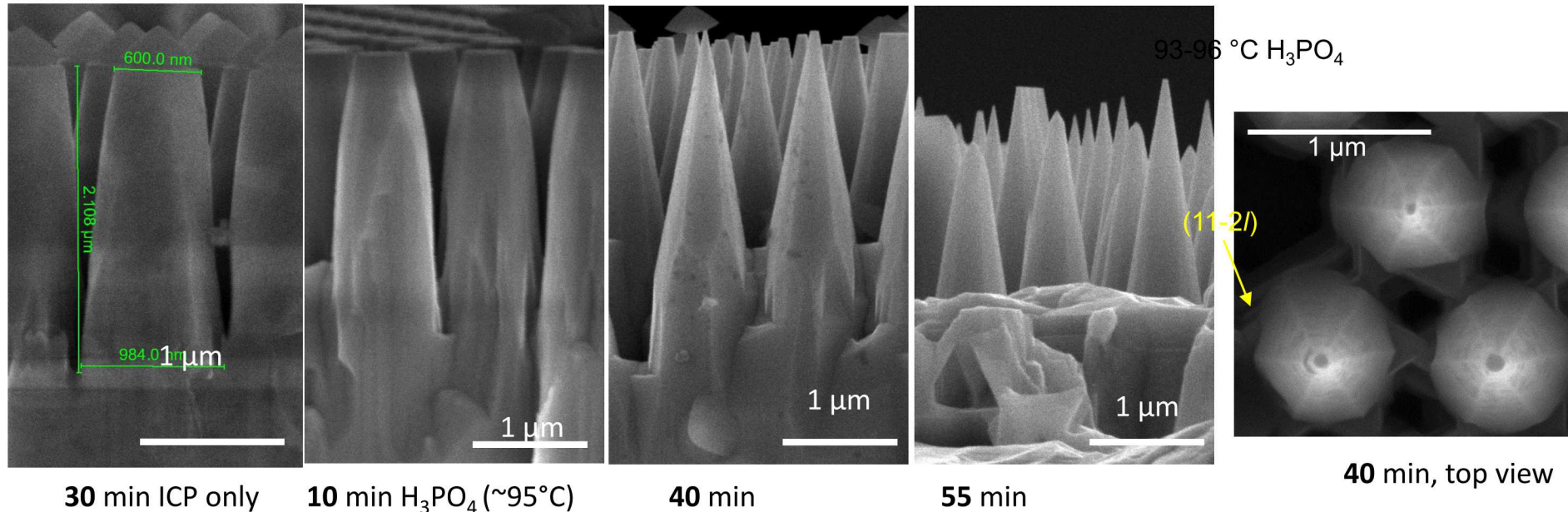
# Top-down fabrication of AlGaN nanowires?



65% nAlGa<sub>N</sub>, 65°C AZ400K

- Preliminary results show initial etch similar to GaN, with the same 'm-' ( $1\bar{1}00$ ) plane type micro-faceting
- Etch pits aligned with 'm-' ( $1\bar{1}00$ ) plane
- However, no prominent GaN-like vertical/lateral 'steps' observed during etch duration
- Further work needed to improve or explain etch

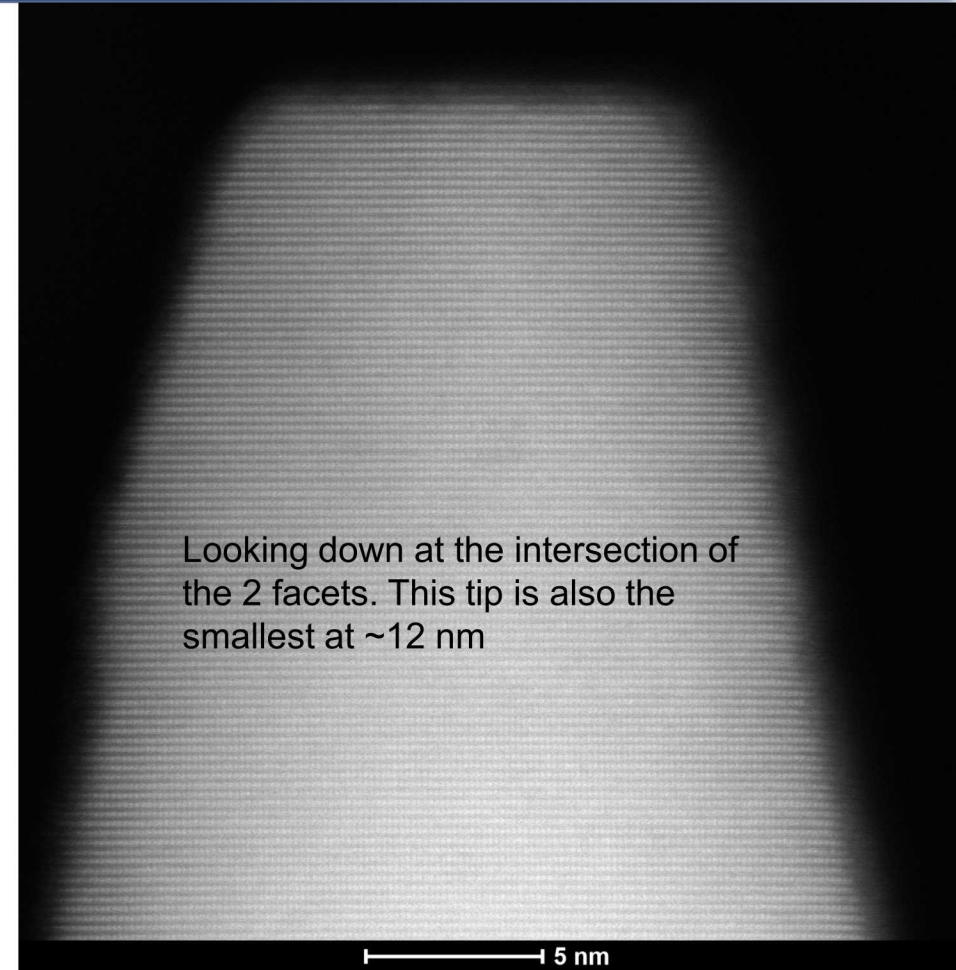
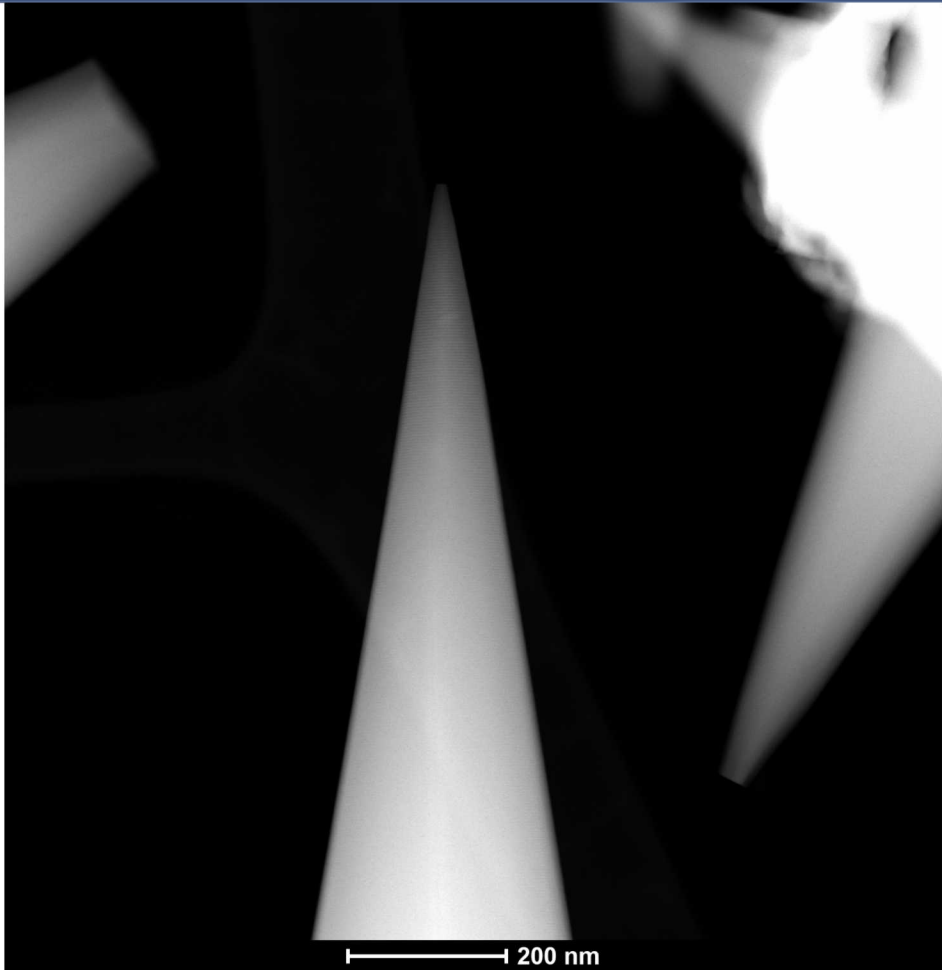
# $\text{H}_3\text{PO}_4$ wet etching of tapered GaN nanowires



- $\text{H}_3\text{PO}_4$  wet etching of ICP dry etched GaN nanowires leads to inclined {11-2/} facets not seen in KOH-based etch
- Micro-faceting not observed during etch in contrast to KOH-based etch. Also top corners not “protected” as in KOH-based etch.
- Leads to “pointy” tapered nanowires instead of straight vertical nanowires

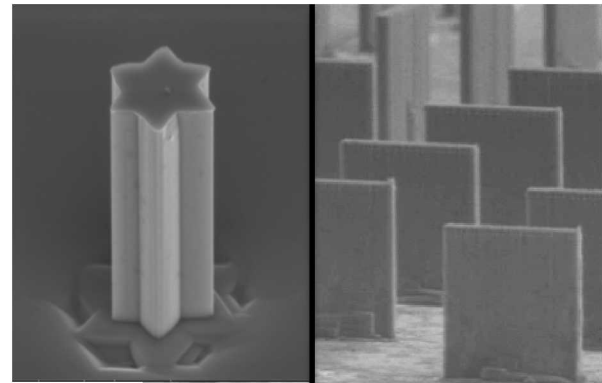
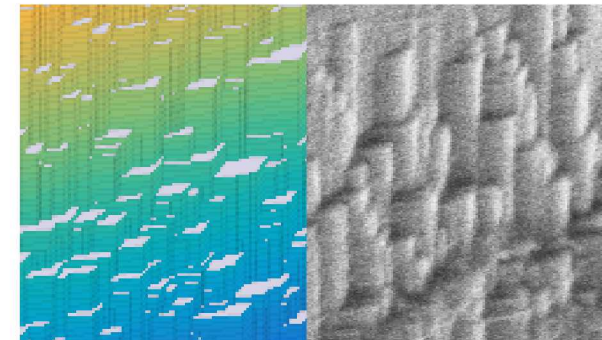
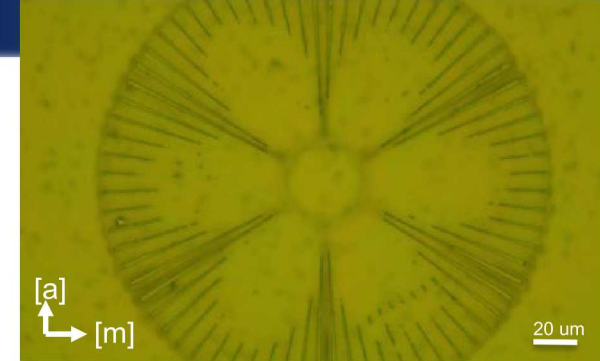


# STEM - $\text{H}_3\text{PO}_4$ etched nanowires



# Conclusions

- **Wet etching: ideal technique for III-nitride nanostructure fabrication due to extremely high anisotropy**
  - Wulff-Jaccodine: **prediction of pillar etch geometry and facets**
  - **Mechanism for facet evolution** from tapered to straight
  - Phosphoric acid vs. KOH -> different facets.  
Possibility for other, different 3D wet etches?
- **Complex cross-sectional geometries** with atomic-scale sidewall smoothness, for **III-nitride based nano-devices**





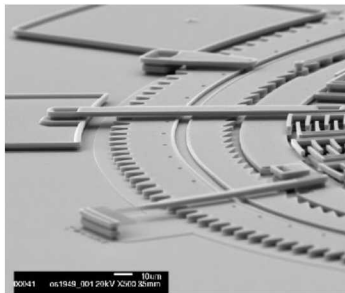
# BACKUP SLIDES

# Problem: Lack of 3D nano-microfabrication processes for III-nitrides and III-V compound semiconductors

Control over 3D shape and composition at nano-microscale can enable *manipulation of properties, new physics and functionalities, and enhanced performance.*

## 3D Silicon structures

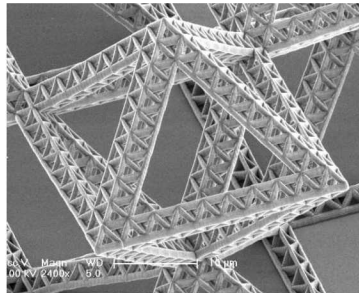
N/MEMS



Torsional Ratcheting  
Actuator

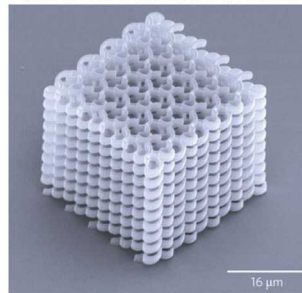
SNL

Nanolattice/truss



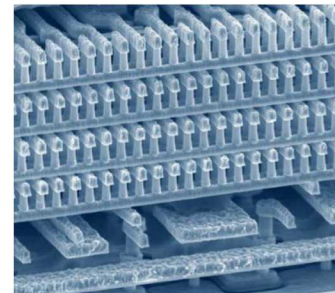
J. Greer, Caltech (2014)

3D Photonic Crystals  
and Metamaterials



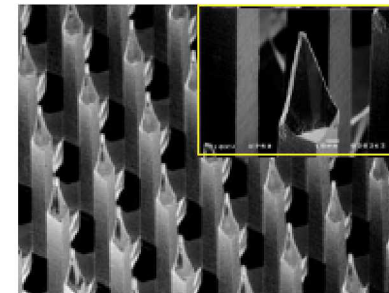
Farsari et al. *Nat. Phot.* (2009)

3D FinFET/NAND



Matrix Semiconductor (2012)

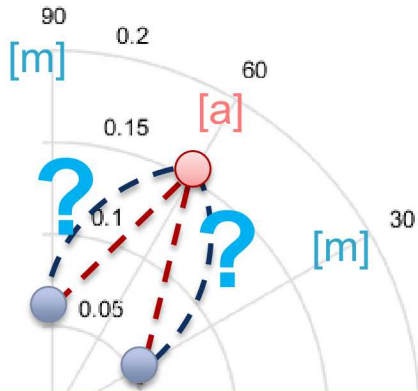
3D Nanoneedles



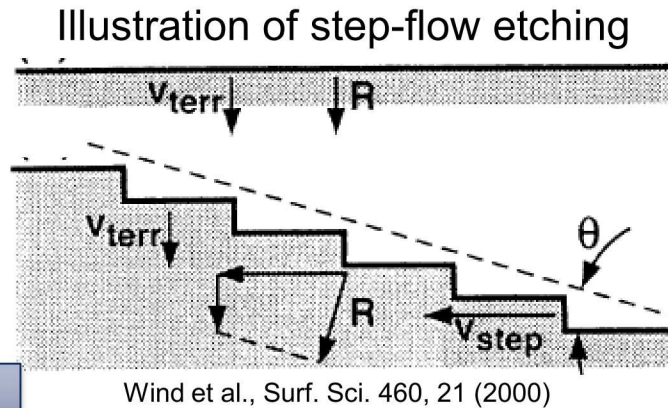
Shikida et al. (2004)

Methods to realize tailored 3D nanostructures are significantly underdeveloped for GaN and even classic III-V semiconductors (e.g. GaAs), hindering development of novel and next generation nano- and micro- enabled semiconductor structures and devices for solid-state lighting, high-speed and high power computer, quantum emitters, energy harvesting,, etc.

# How Does the Etch Rate Depend on Orientation?



What is the shape of this curve?



Atomically flat facet,  
e.g. m and/or a plane

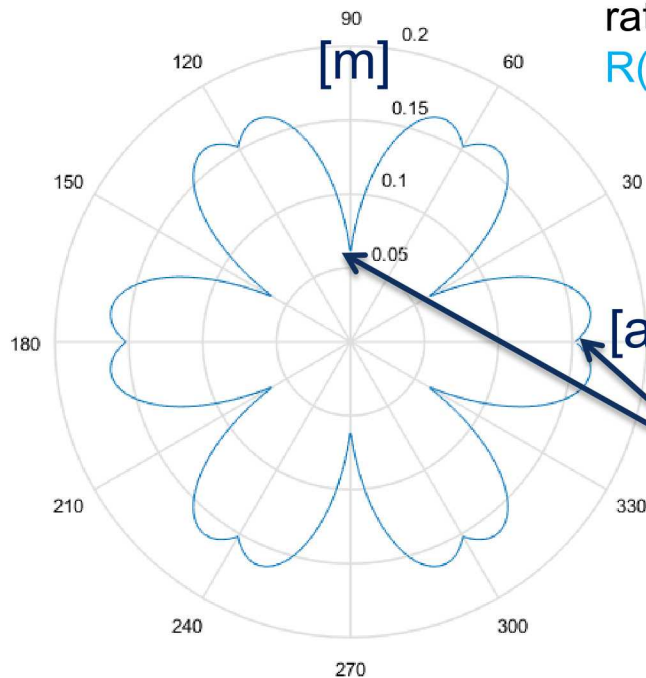
Misoriented facet with offcut  $\theta$ ,  
e.g. facets between m and a  
planes

Between low index facets (i.e. with zero steps), the etch rate as a function of orientation is:

$$R(\theta) = V_{\text{terr}} \cos(\theta) + V_{\text{step}} \sin(|\theta|)$$

Expected form for orientation dependent etch rates, given m and a etch rates, and that **both m and a are atomically flat surfaces**

$$R(\theta) = m \cos(\theta\pi/(\pi/3)) + a \sin(|\theta\pi/(\pi/3)|), \text{ for } \theta: -\pi/6 < \theta < \pi/6$$

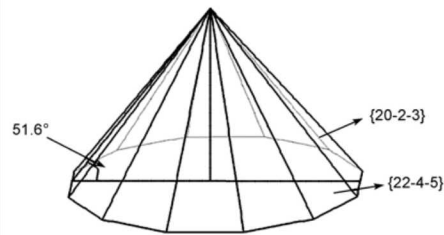
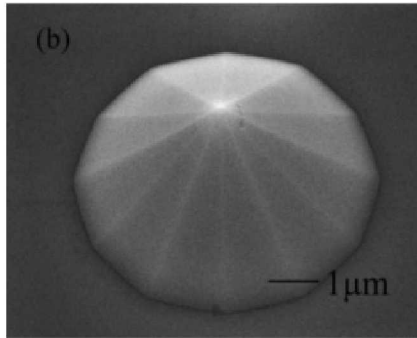


Local minima for atomically flat surfaces –  
as is for the growth Wulff plot



# Prior $\text{H}_3\text{PO}_4$ Wet etching of planar GaN

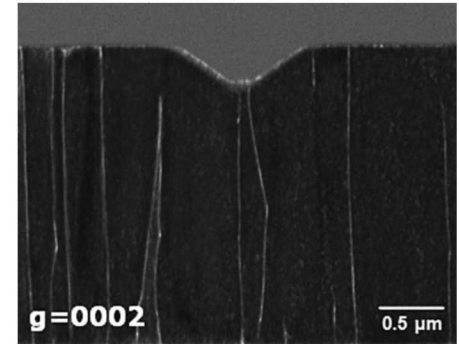
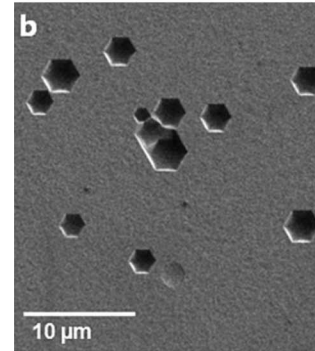
## N-polar GaN



Qi et al., Sci. China, Tech. Sci., 53, 769 (2010)

- N-face GaN etched in  $\text{H}_3\text{PO}_4$  (120°C) 5 minutes yields dodecagonal pyramids centered around dislocations
- Etching preferentially slows down on  $(22\bar{2}3)/(22\bar{4}5)$  semipolar planes due to low surface energies

## Ga-polar $\text{Al}_{0.22}\text{GaN}$ (20 nm)/GaN

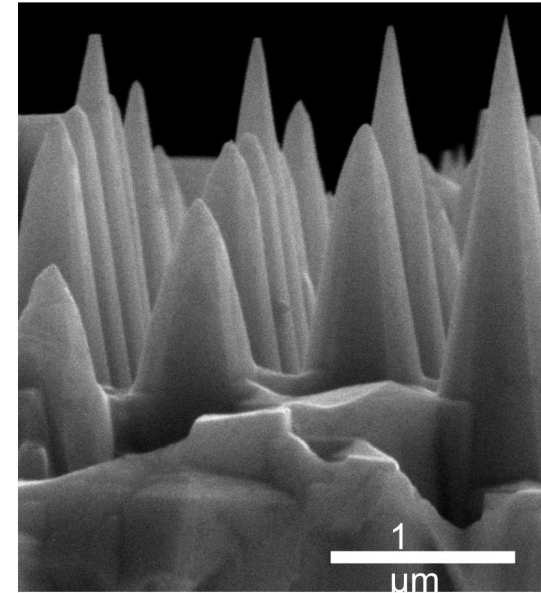
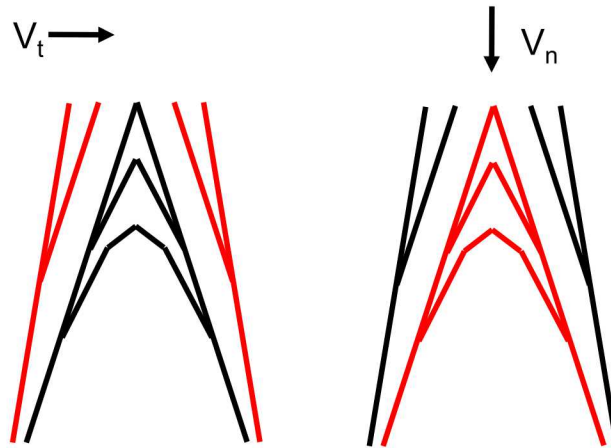


M. Reiner et al., PSSB, 252, 1121 (2015)

- Ga-face GaN etched in  $\text{H}_3\text{PO}_4$  (150-245°C) yields hexagonal pits centered around dislocations with semipolar  $\{101k\}$  or  $\{112l\}$  facets

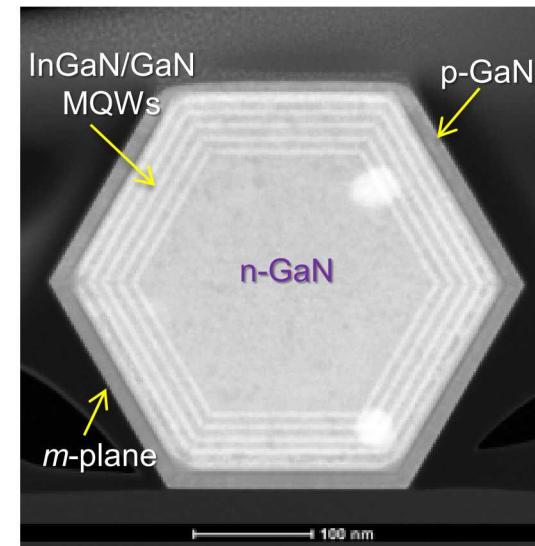
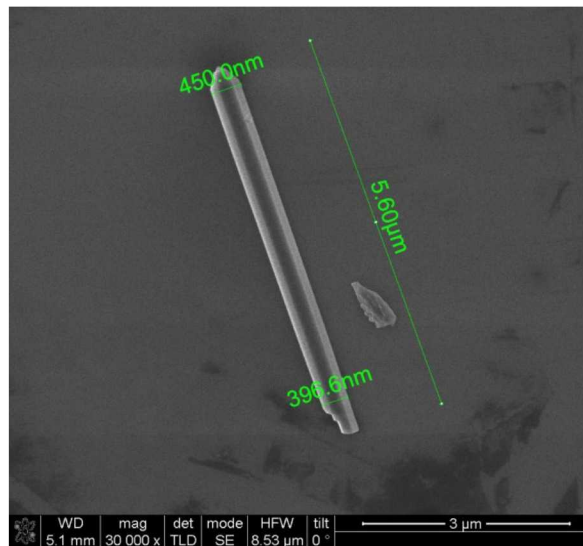
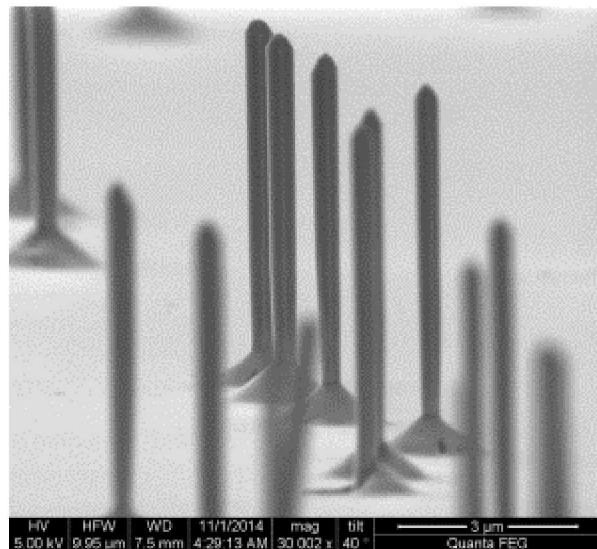
# Nanowire Etch Progression Hot $\text{H}_3\text{PO}_4$

The etch appears to move tangentially and then normally to the Ga surface



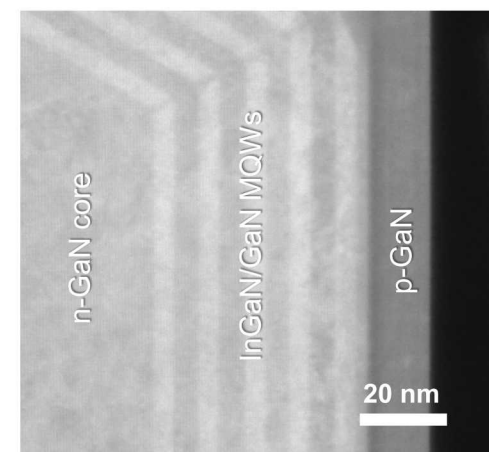
Semipolar facets form due to  $V_n/V_t$  ratios. Initially,  $V_t$  dominates where it is dominated by concentration (kinetics). When  $V_n$  dominates, an increase in semipolar angle occurs, and this is driven more by temperature and surface energy (thermodynamics).

# Hybrid top-down bottom-up fabrication of core-shell p-i-n InGaN/GaN core-shell NWs



*Core-shell radial p-i-n InGaN/GaN MQW nanowires following MOCVD regrowth on top-down fabricated n-type GaN nanowires*

- Regrowth on top-down n-GaN nanowire results in taper. Growth of semipolar planes also results in non-flat tip, which may lower mirror reflectivity
- Growth conditions, layer thicknesses, varied over several runs, then nanowires tested by optical pumping for lasing

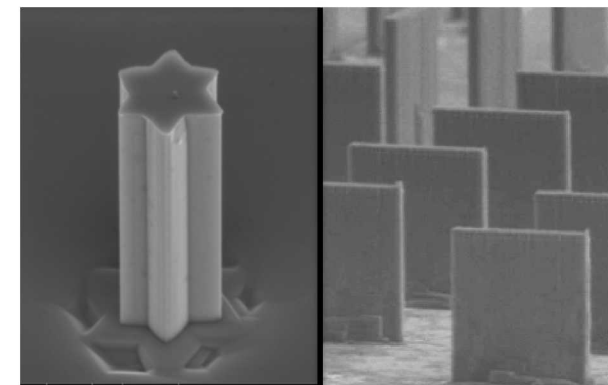
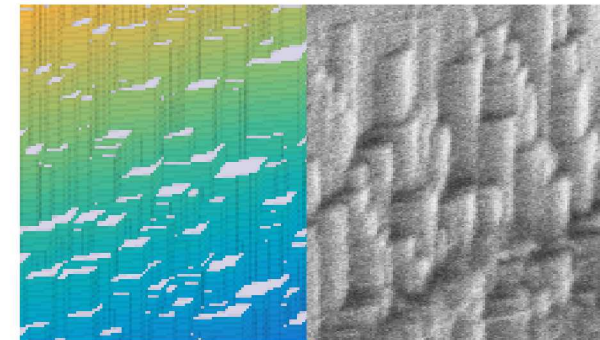
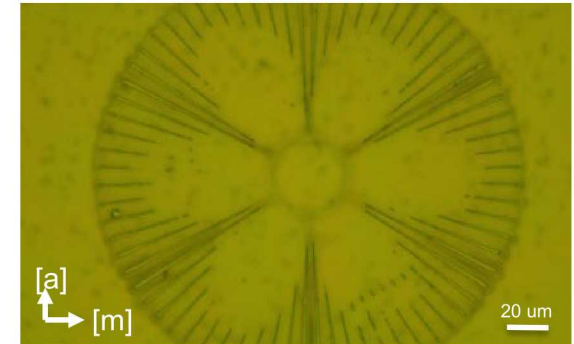


STEM by Ping Lu

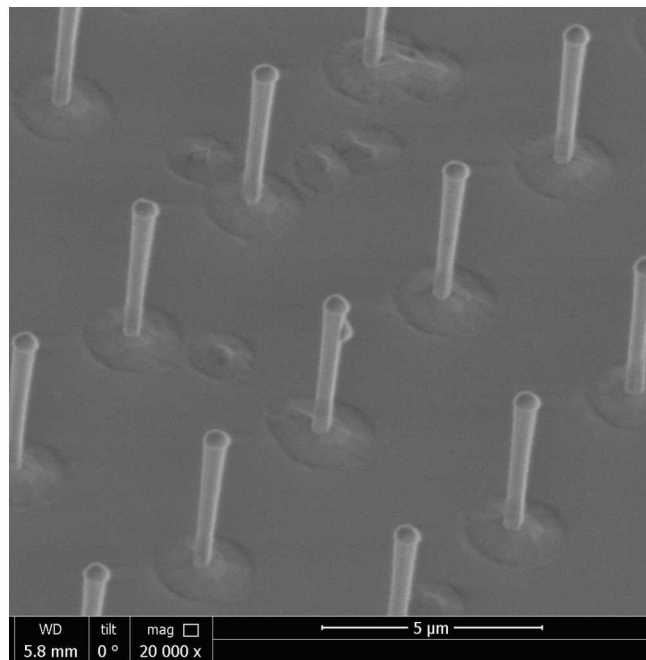


# Summary – Top-down fabrication of III-nitride nanostructures for photonics

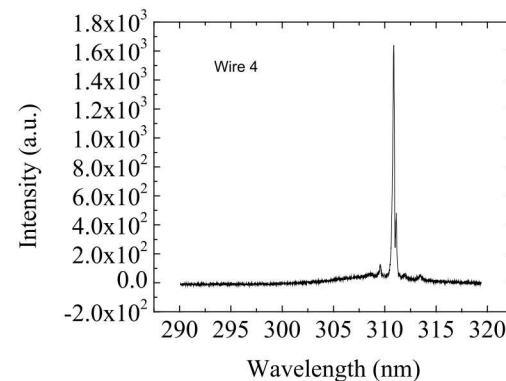
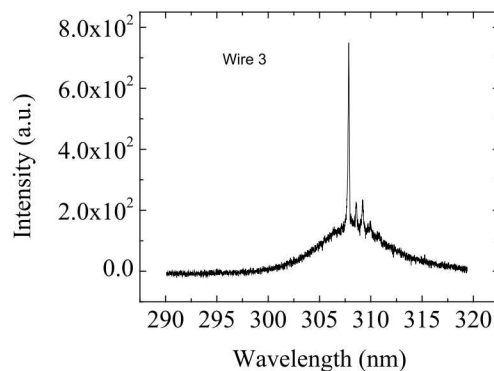
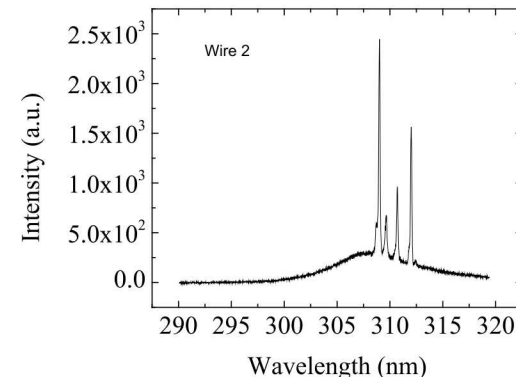
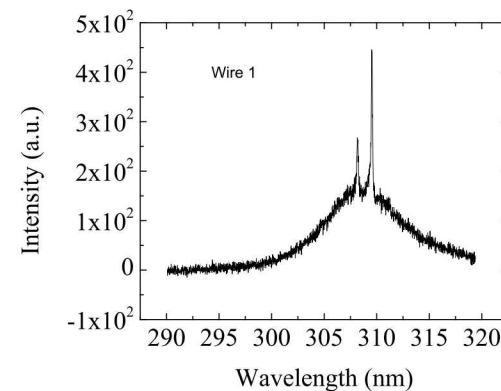
- **First direct orientation resolved wet etch rate measurements** for the nonpolar GaN set of planes
- Implementation of Wulff-Jaccodine method for **prediction of pillar etch geometry and facets**
- **Elucidated mechanism for facet evolution** from tapered to straight nanostructures
- Demonstrated **complex cross-sectional geometries** for vertical nanostructures with possible atomic-scale sidewall smoothness
- Demonstrated **mode, polarization, beam-shape, and wavelength control** in top-down etched nanowire lasers
- **Wet chemical etching is an ideal technique for high-quality III-nitride nanostructure fabrication due to its extremely high anisotropy**



# Top-down AlGaN nanowires - lasing achieved

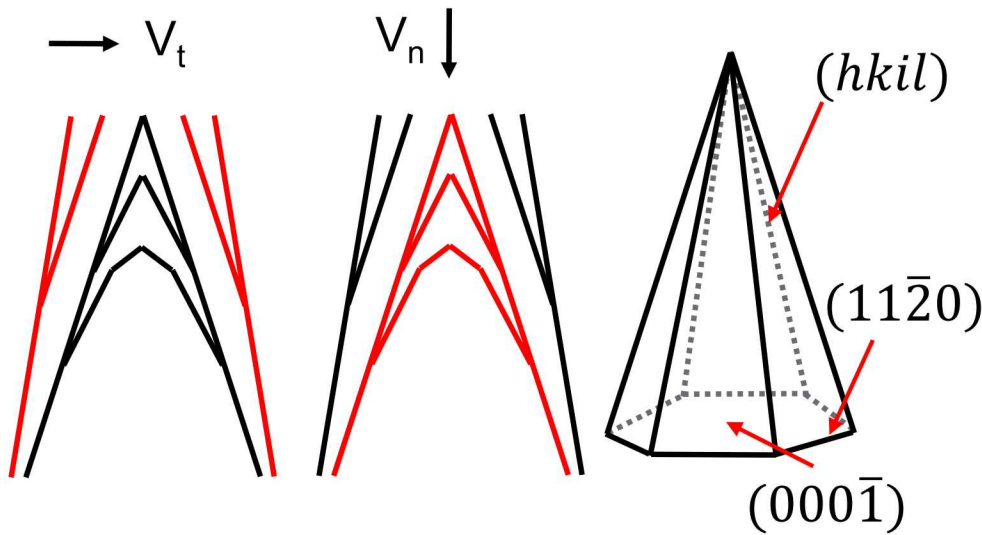
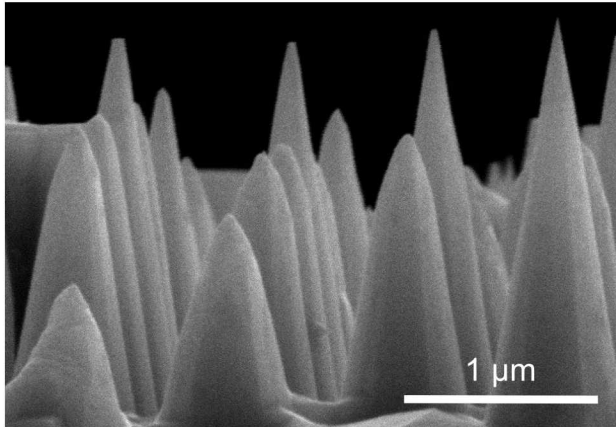


Nanowire Length: 4.0 – 4.1 μm  
Diameter: Top ~420-440 nm, bottom: 350-380 nm



- Single top-down  $\text{Al}_{0.31}\text{Ga}_{0.69}\text{N}$  nanowires show multimode lasing ~308-312 nm
- Estimated threshold ~200 kW/cm<sup>2</sup> (similar to top-down GaN nanowires)
- *Nanowires burn-out after only ~3-6 seconds* (less resilient than GaN nanowires!?)

# $\text{H}_3\text{PO}_4$ nanowire wet etching evolution

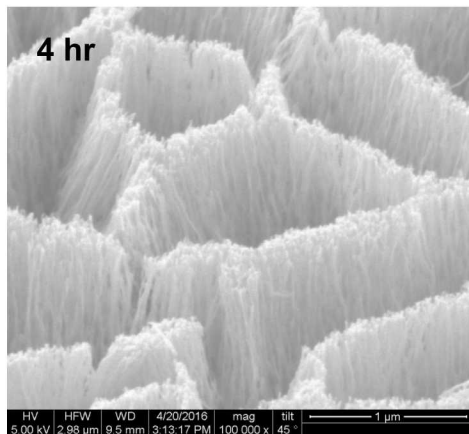
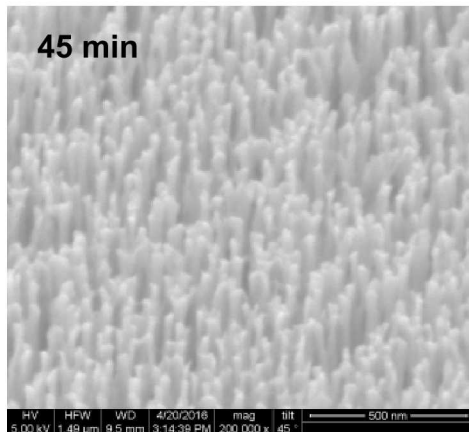


- High-index semi-polar planes are etched
- Semi-polar facets form due to  $V_n/V_t$  ratios
- $V_t$  initially dominates  $\rightarrow$  driven by concentration (kinetics)
- $V_n$  driven by surface energy (thermodynamics)
- Semi-polar  $(hkil)$  plane progression:  
 $73 - 82^\circ \sim (11\bar{2}1) \rightarrow 60^\circ \sim (11\bar{2}2) \rightarrow 50 - 45^\circ \sim (11\bar{2}3)$
- Preliminary results support sequential increase of ' $l$ ' index

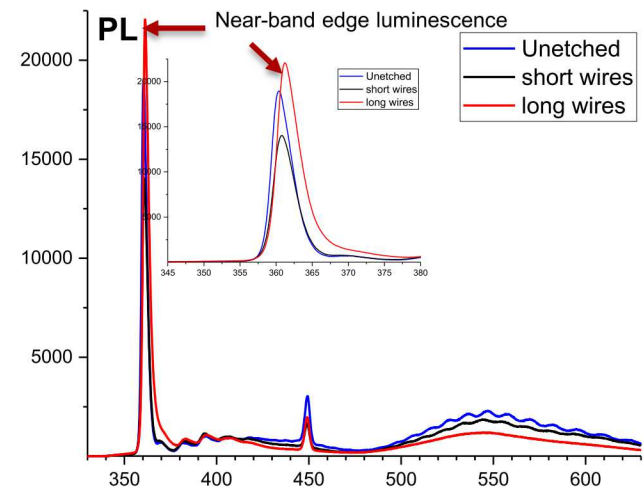
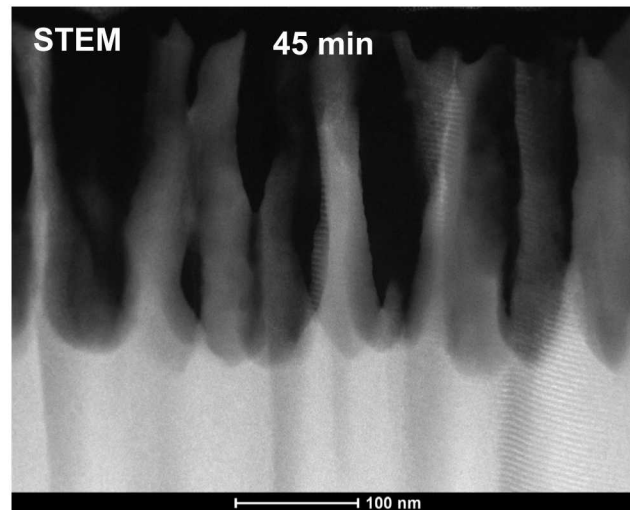


# PEC etched GaN nanowires with very high aspect-ratios

PEC etch: 370 nm, 4 mW,  
1V, 2M H<sub>2</sub>SO<sub>4</sub>



- GaN epilayer PEC etched at 370 nm results in dense array of thin GaN nanowires
- Cross-sectional STEM shows single-crystalline GaN nanowires; not related to dislocation cores, as previous reports showed
- Vertical etch through entire GaN film with little diameter change – self limiting lateral etch – near-surface carrier depletion effects?
- Despite high surface/volume, PEC etched NWs show comparable or better optical properties compared to unetched GaN film

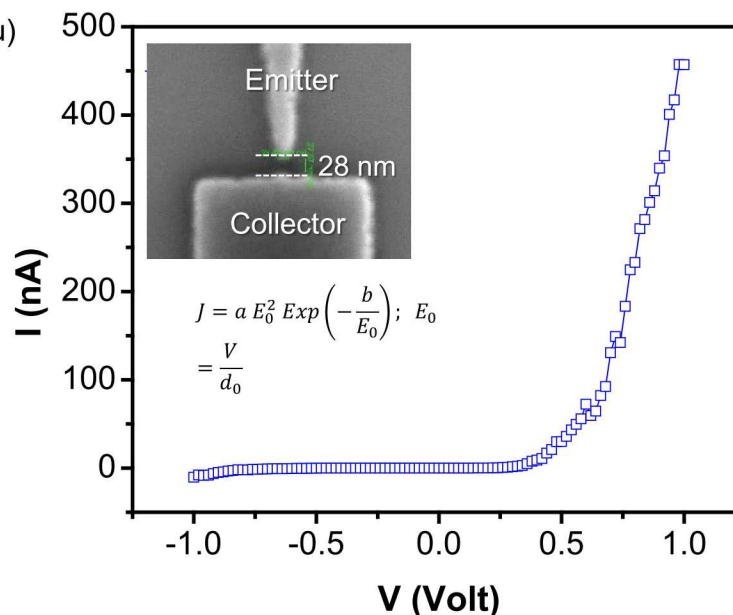
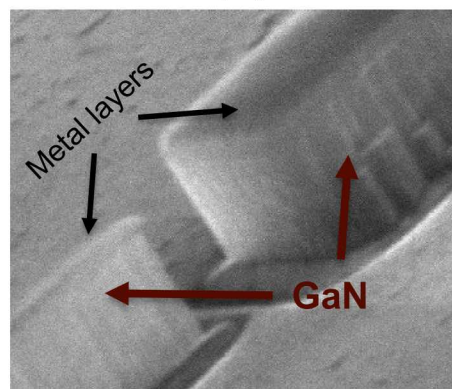
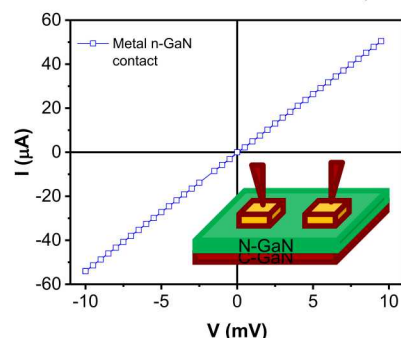


# Working GaN field emission devices (nanodiodes)!

**Proof-on-concept: Successful field emission (FE) in air with low turn on voltage and high emission current!**

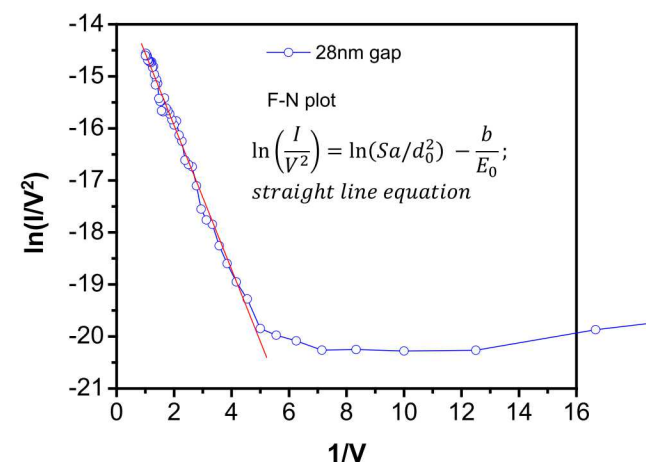
Field emission is diode-like for sharp emitter and flat collector (expected)

Ohmic contacts to n-GaN (Ti/Ni/Au)



IV measurement: **Diode** characteristics

**“Hero” device shows very low turn on and very high FE current!**



**Fowler-Nordheim (FN) test for field emission**

*Straight line fitting of data*

*Slope = -1.40; Y-intercept = -13.16*

**Field enhancement factor  $\beta$  = 920 using  $\phi = 4 \text{ eV}$**