

Discharge Mechanism of the $\gamma\text{-MnO}_2$ Electrode in Shallow-Cycled Zn/MnO₂ Batteries: An Ab Initio Study

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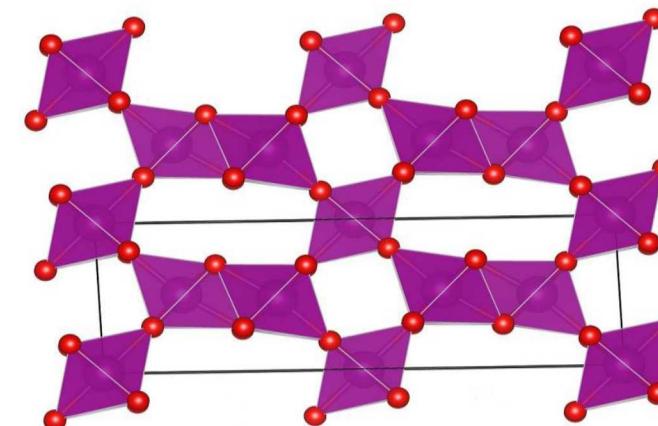
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Outline

- **Background and Motivation**
- **Computational Methods**
- **Hydrogen insertion in $\gamma\text{-MnO}_2$**
- **Summary**

$\gamma\text{-MnO}_2$ “intergrowth”



Background and Motivation

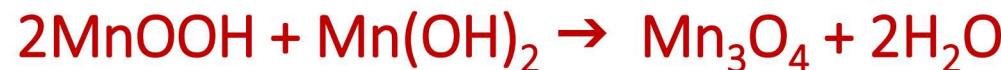
- Alkaline Zn/MnO₂ batteries hold great promise for electrical energy storage and load-leveling power grid applications due to their high energy density, non-toxicity, and low cost.
- Development of commercially viable rechargeable Zn/MnO₂ alkaline batteries has been hindered by a short cycle life due to irreversible changes occurring in the MnO₂ cathode and Zn anode.
- Theoretical studies have been focused on the development of a computational model for accurate prediction of the electrochemical behavior of γ -MnO₂ in Zn/MnO₂ batteries.
- The main objective of this research is to analyze the relationship between the depth of discharge (DOD) and the probability of formation of irreversible redox reaction products in the γ -MnO₂ cathode.

Cathode Half-Cell reaction in Zn/MnO₂ Batteries

Electrochemical reaction (γ -MnO₂ cathode)



Extensive formation of Mn(OH)₂ leads to further conversion of cathode to other forms such as hausmannite



Initial MnO₂ discharge reaction is the proton insertion in solid phase of γ -MnO₂



Computational Methods

- Calculations are done using an *ab initio* computational methods based on density functional theory (**DFT**).

Kohn-Sham equations:

$$\left(-\frac{\nabla^2}{2} + V_{eff}[\mathbf{n}](\mathbf{r}) \right) \psi_i(\mathbf{r}) = \epsilon_i \psi_i(\mathbf{r})$$

$$V_{eff}[\mathbf{n}](\mathbf{r}) = V_{ext}(\mathbf{r}) + \int \frac{\mathbf{n}(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d^3\mathbf{r}' + V_{xc}[\mathbf{n}](\mathbf{r})$$

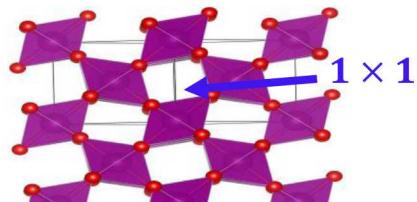
- **Quantum ESPRESSO*** (opEn-Source Package for Research in Electronic Structure, Simulation, and Optimization) code for electronic structure calculations.
- Exchange correlation functional was approximated by GGA PBEsol.
- Vanderbilt ultra- soft pseudopotentials**
- Spin polarized calculations.

* <http://www.quantum-espresso.org>

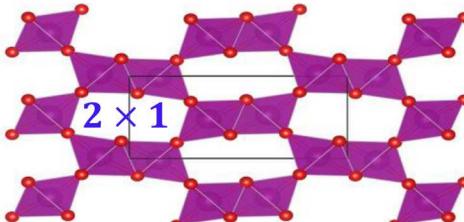
**<http://www.physics.rutgers.edu/~dhv/uspp/>

Stability of MnO_2 polymorphs

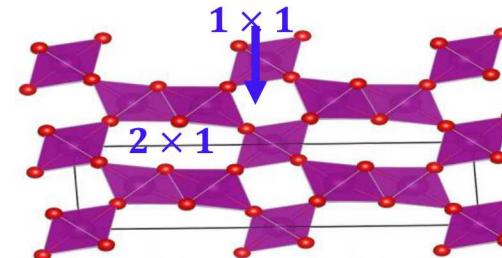
- $\beta\text{-MnO}_2$ is supposed to be the lowest energy structure



Pyrolusite (β)

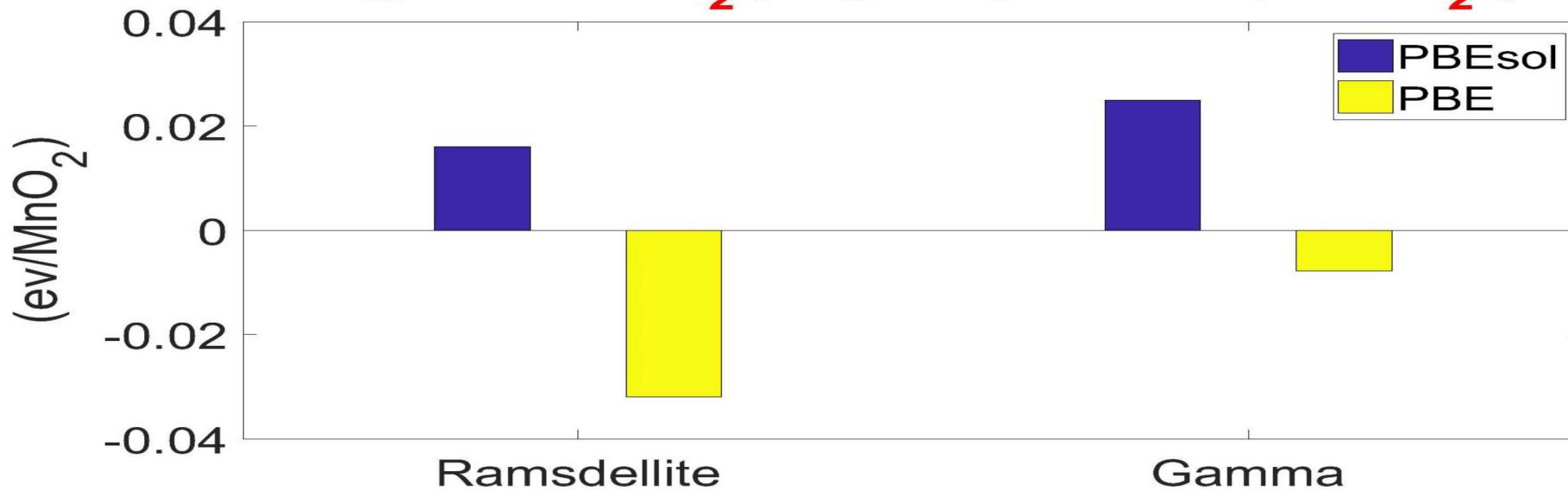


Ramsdellite (R)



Intergrowth (γ)

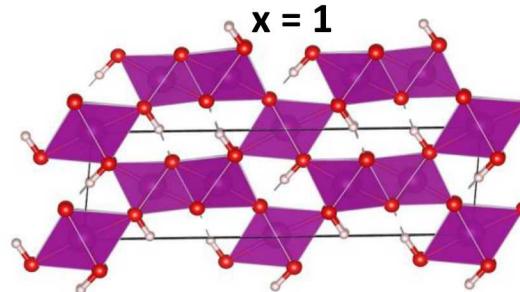
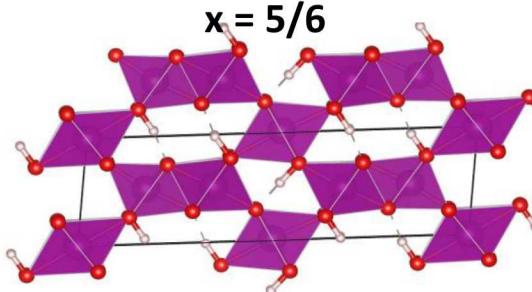
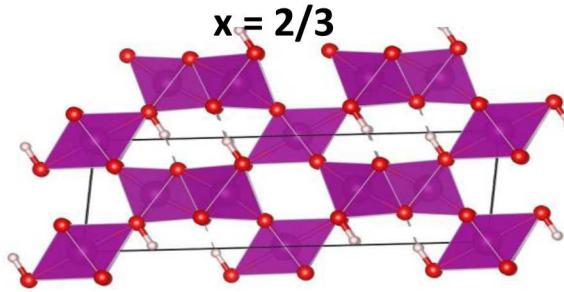
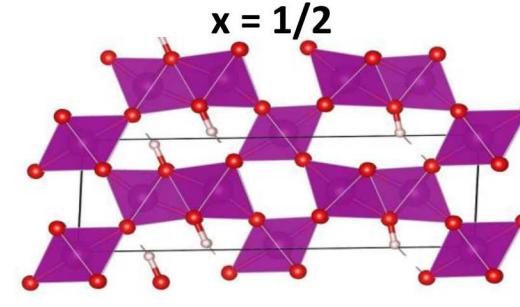
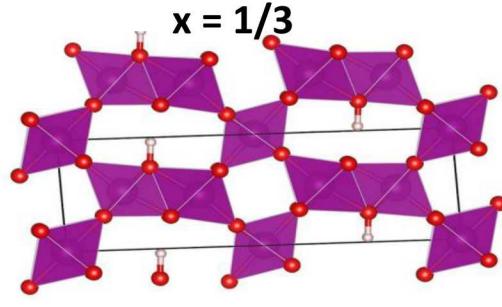
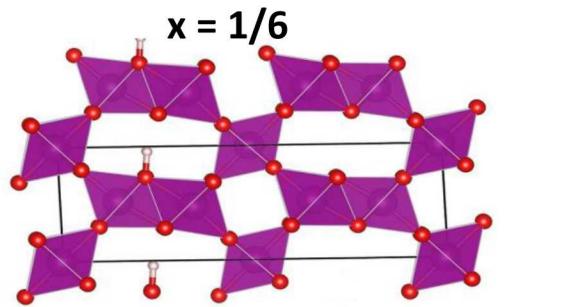
Energies of MnO_2 polymorphs w.r.t $\beta\text{-MnO}_2$ (ev)



- GGA PBEsol xc functional predicts the correct order of the energies of MnO_2 polymorphs

Hydrogen insertion in $\gamma\text{-MnO}_2$

Calculated Lowest Energy Structures of $\text{MnO}_{2-x}(\text{OH})_x$ for $0 \leq x \leq 1$



- Protonation is carried out in three stages:
 - (1) One H atom is inserted in each 2x1 tunnel,
 - (2) Two H atoms are inserted in each 2x1 tunnel,
 - (3) One H atom is inserted in each 1x1 tunnel.
- Protonation produces significant structural distortions in $\gamma\text{-MnO}_2$.

Hydrogen insertion in γ -MnO₂

Energy and volume changes of γ -MnO₂ with increase in DOD

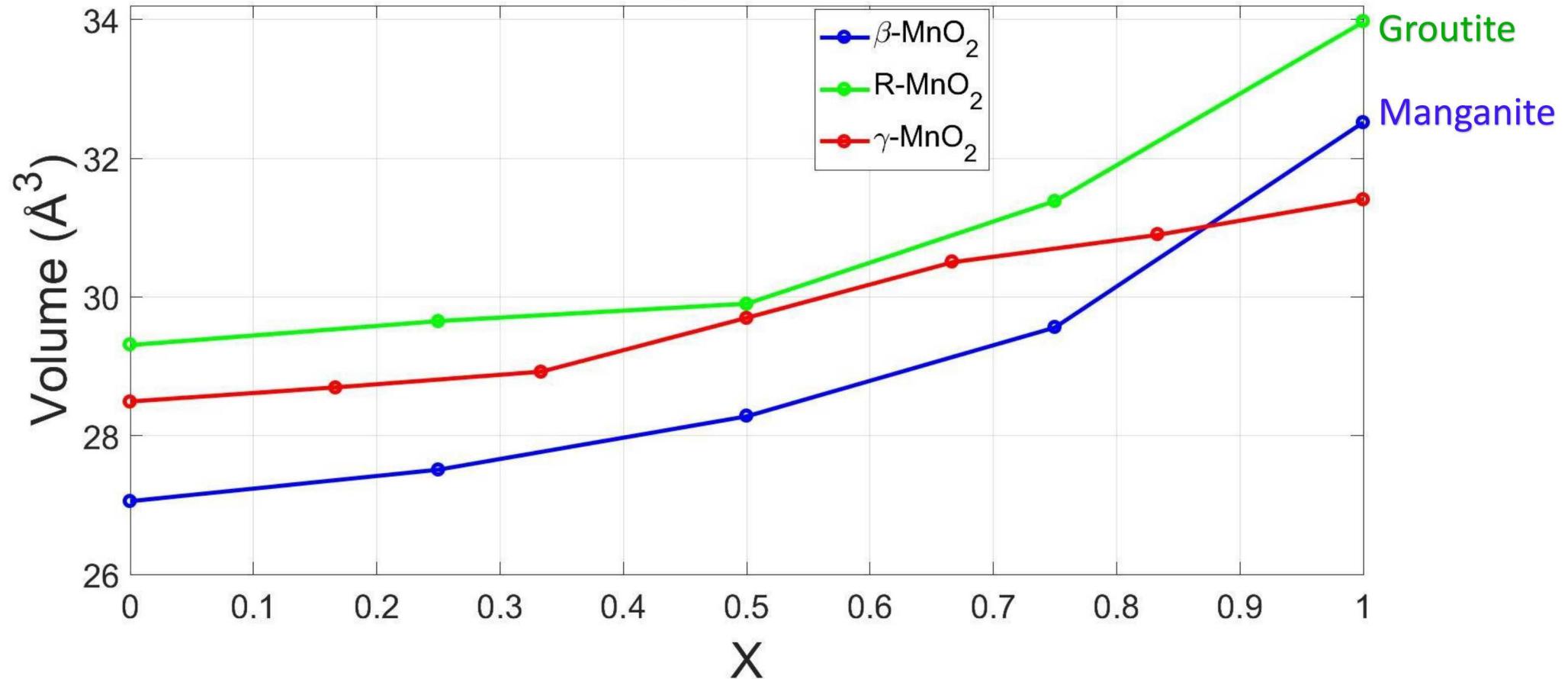
S.N	Structure	ΔE (ev)	$\Delta E/H$ (ev)	$\text{\AA}^3/\text{MnO}_2$	$\Delta V/\text{MnO}_2$ (%)
1	Mn ₆ O ₁₂	0	0	28.4864	0
2	Mn ₆ O ₁₂ H	-0.9816	-0.9816	28.6923	0.72
3	Mn ₆ O ₁₂ H ₂	-1.9402	-0.9701	28.9220	1.53
4	Mn ₆ O ₁₂ H ₃	-2.5245	-0.8415	29.6950	4.24
5	Mn ₆ O ₁₂ H ₄	-3.2899	-0.8225	30.5028	7.08
6	Mn ₆ O ₁₂ H ₅	-3.4837	-0.6967	30.8996	8.47
7	Mn ₆ O ₁₂ H ₆	-3.6667	-0.6111	31.4080	10.26

$$\Delta E = E(\text{Mn}_6\text{O}_{12}\text{H}_n) - E(\text{Mn}_6\text{O}_{12}) - \frac{n}{2} E(\text{H}_2) ; n = 1, 2, \dots, 6$$

$$\Delta V = [V(\text{Mn}_6\text{O}_{12}\text{H}_n) - V(\text{Mn}_6\text{O}_{12}\text{H}_{n-1})] / V(\text{Mn}_6\text{O}_{12}\text{H}_{n-1}) ; n = 1, 2, \dots, 6$$

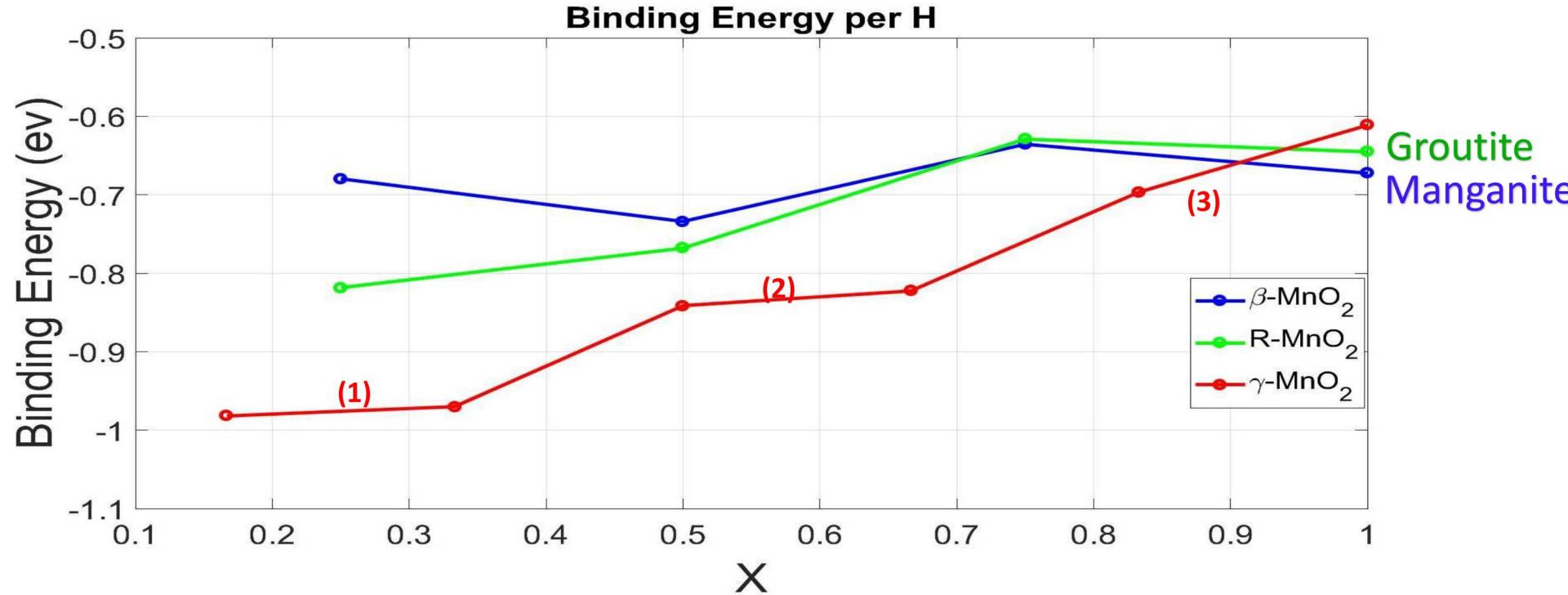
- Energy of H-insertion is lower for 2x1 tunnels than for 1x1 tunnels.

Volume/ MnO_2 with increasing DOD



- Initial volumes: $V_\beta < V_\gamma < V_R$
- Volume of protonated MnO_2 phases increases nonlinearly with increasing DOD.
- For large DOD volume of protonated $\beta\text{-MnO}_2$ becomes larger than volume of protonated $\gamma\text{-MnO}_2$

Binding energy comparison with other forms of MnO_2



- For protonated $\gamma\text{-MnO}_2$ binding energy per H atom decreases significantly with increasing DOD.
 $|E_b^{\text{stage}1}| > |E_b^{\text{stage}2}| > |E_b^{\text{stage}3}|$
- Protonated $\gamma\text{-MnO}_2$ is more stable for small value of X and becomes less stable for large X than other forms of MnO_2
- Protonation of 1x1 tunnels may leads to structural breakdown of $\text{MnO}_{2-x}(\text{OH})_x$

Summary

- Energy of H-insertion is lower for 2x1 R- MnO_2 tunnels than for 1x1 $\beta\text{-MnO}_2$ tunnels.
- Protonation produces significant structural distortions in MnO_2 polymorphs.
- Initially, inserted protons occupy 2x1 tunnels of $\gamma\text{-MnO}_2$
- Protons insertion into 1x1 tunnels may lead to breakdown of $\gamma\text{-MnO}_2$.
- Battery life cycle can be extended by limiting protonation to one H atom per 2x1 tunnel.



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