



This paper describes objective technical results and analysis. Any subjective views or opinions that might be expressed in the paper do not necessarily represent the views of the U.S. Department of Energy or the United States Government.

SAND2019-11388C

# Going up in time and length scales in modeling entangled polymers

Gary S. Grest

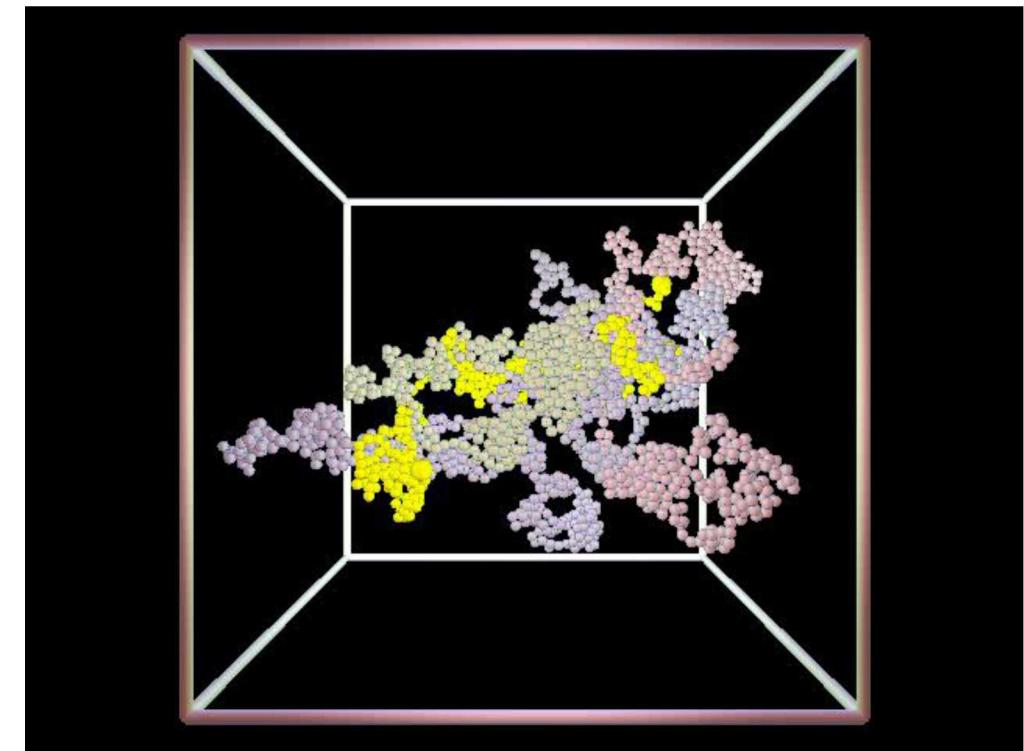
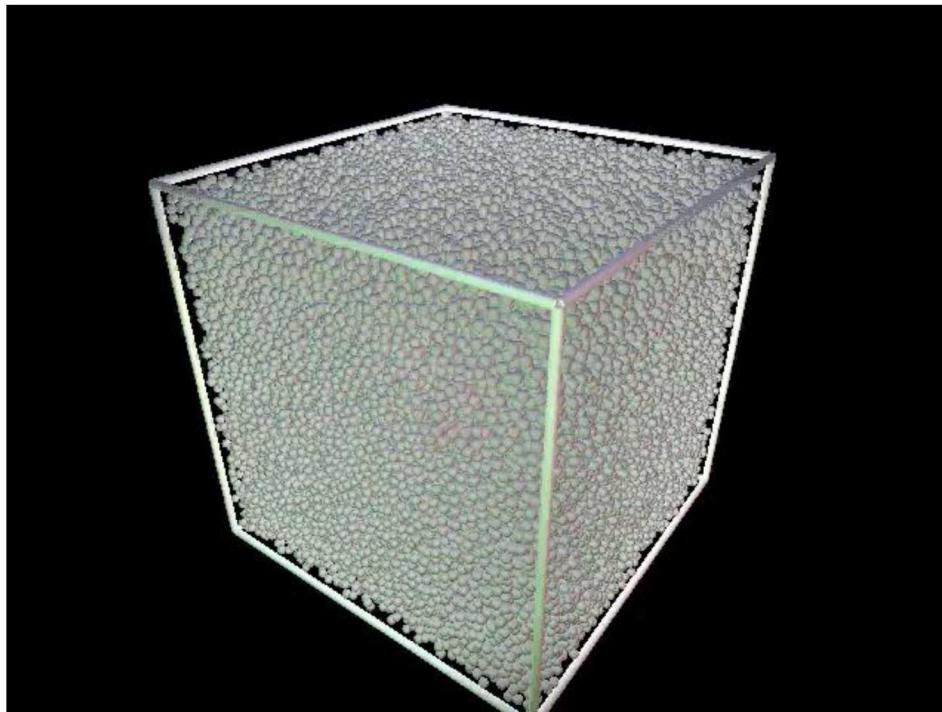
Center for Integrated Nanotechnologies  
Sandia National Laboratories  
Albuquerque, NM

CINT User Meeting  
September 20-22, 2019

This work was performed, in part, at the Center for Integrated Nanotechnologies, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science. Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International, Inc., for the U.S. DOE's National Nuclear Security Administration under contract DE-NA-0003525.

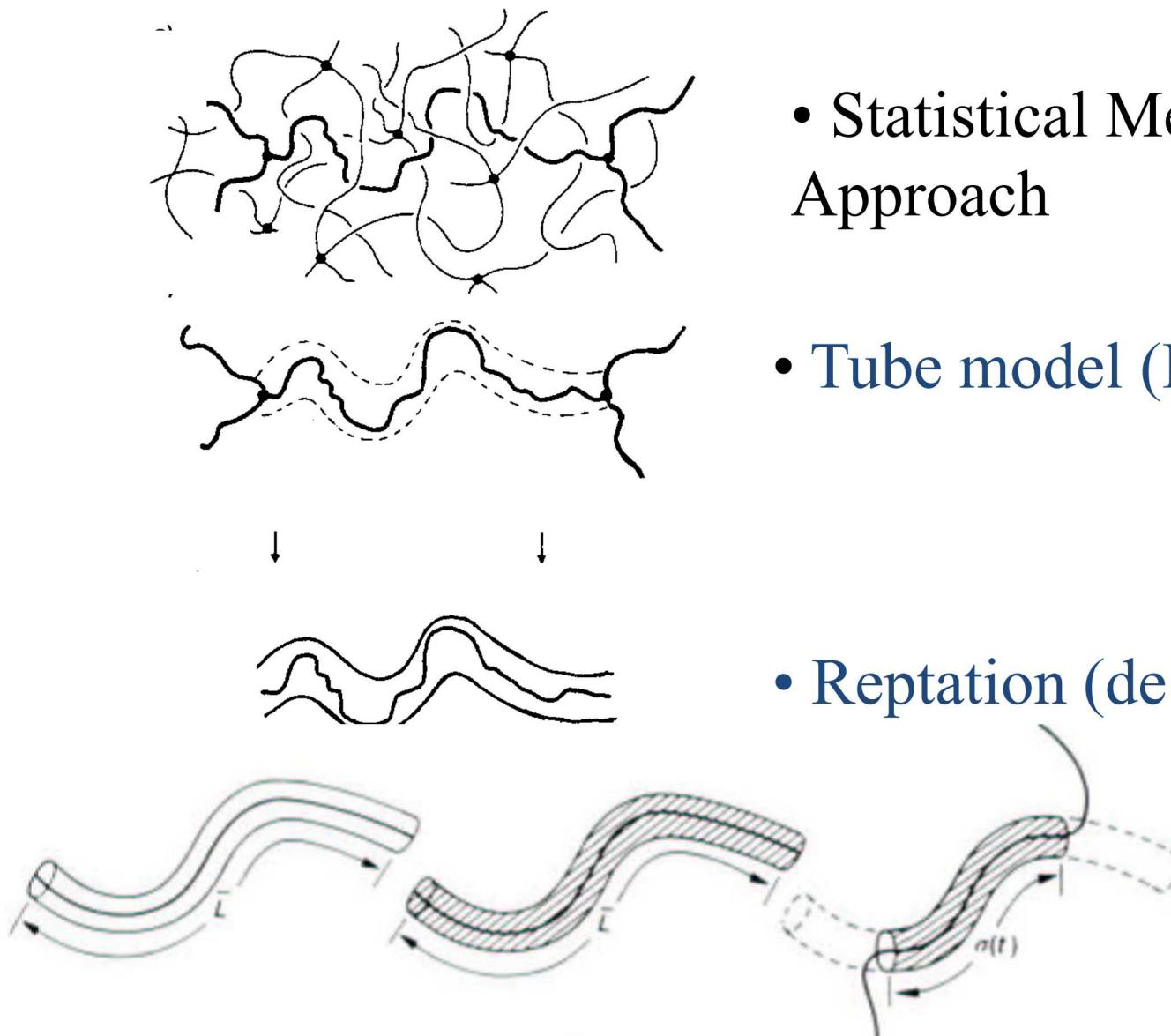
# Polymer Motion

- Polymers are simultaneously hard and soft
  - Unique Viscoelastic Behavior



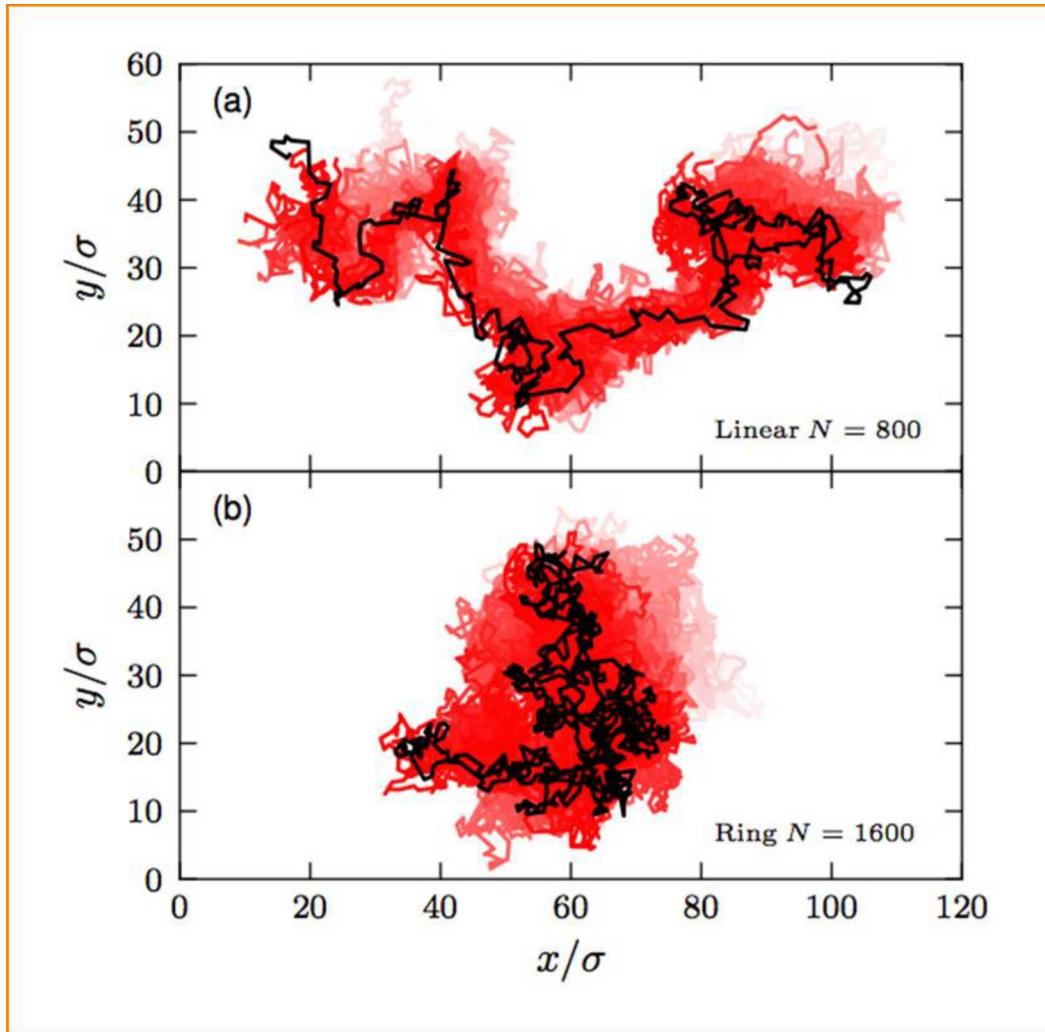
- Motion of a polymer chain is subject to topological constraints

# Polymer Motion in a Liquid



- Statistical Mechanics Approach
- Tube model (Edwards 1967)
- Reptation (de Gennes, 1971)

# Polymer Topology Matters



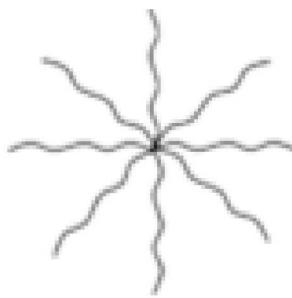
- Linear chains move like a snake
- Rings move by an amoeba-like motion

# Polymer Architectures

Polymers not limited to linear chains



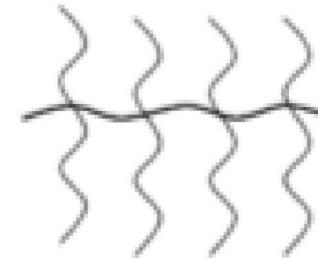
Block copolymer



Star polymer



Comb polymer



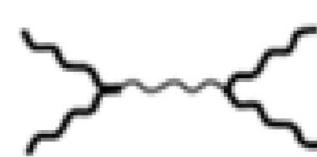
Brush polymer



$AB_2$  star



Palm-tree  $AB_n$



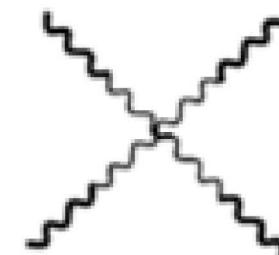
H-shaped  $B_2AB_2$



Dumbbell (pom-pom)



Ring block



Star block  $AB_n$



Coil-cycle-coil



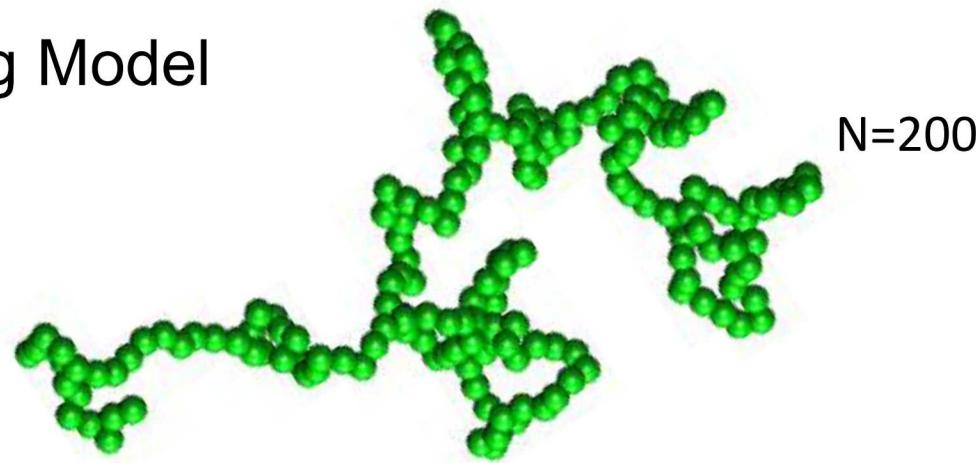
Star  $A_nB_n$

# Computational Challenges in Polymers

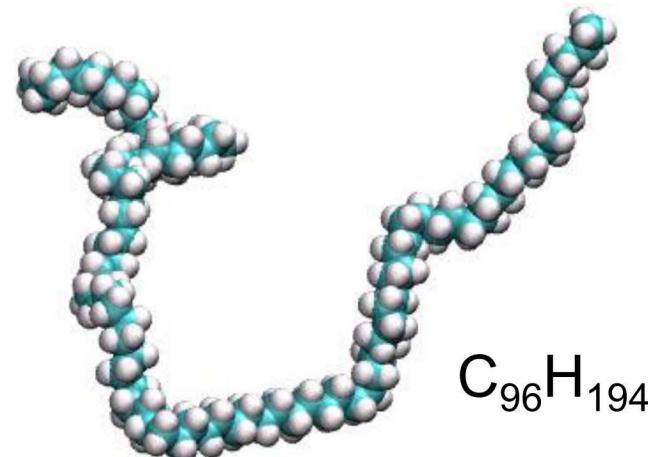
- Longest relaxation time  $\tau \sim N^3$
- Chains are Gaussian coils –  $R \sim N^{1/2}$ 
  - Number of chains must increase as  $R^3 \sim N^{3/2}$  so polymer chains do not see themselves through periodic boundary conditions
- Double chain length – cpu required increases by at least a factor of  $2^{4.5} \sim 23$ 
  - 1-2 month simulation becomes 2-4 years
- Number of processors limited:  $\sim 400-1000$  particles/processor

# Polymer Simulation Models

- Bead-Spring Model

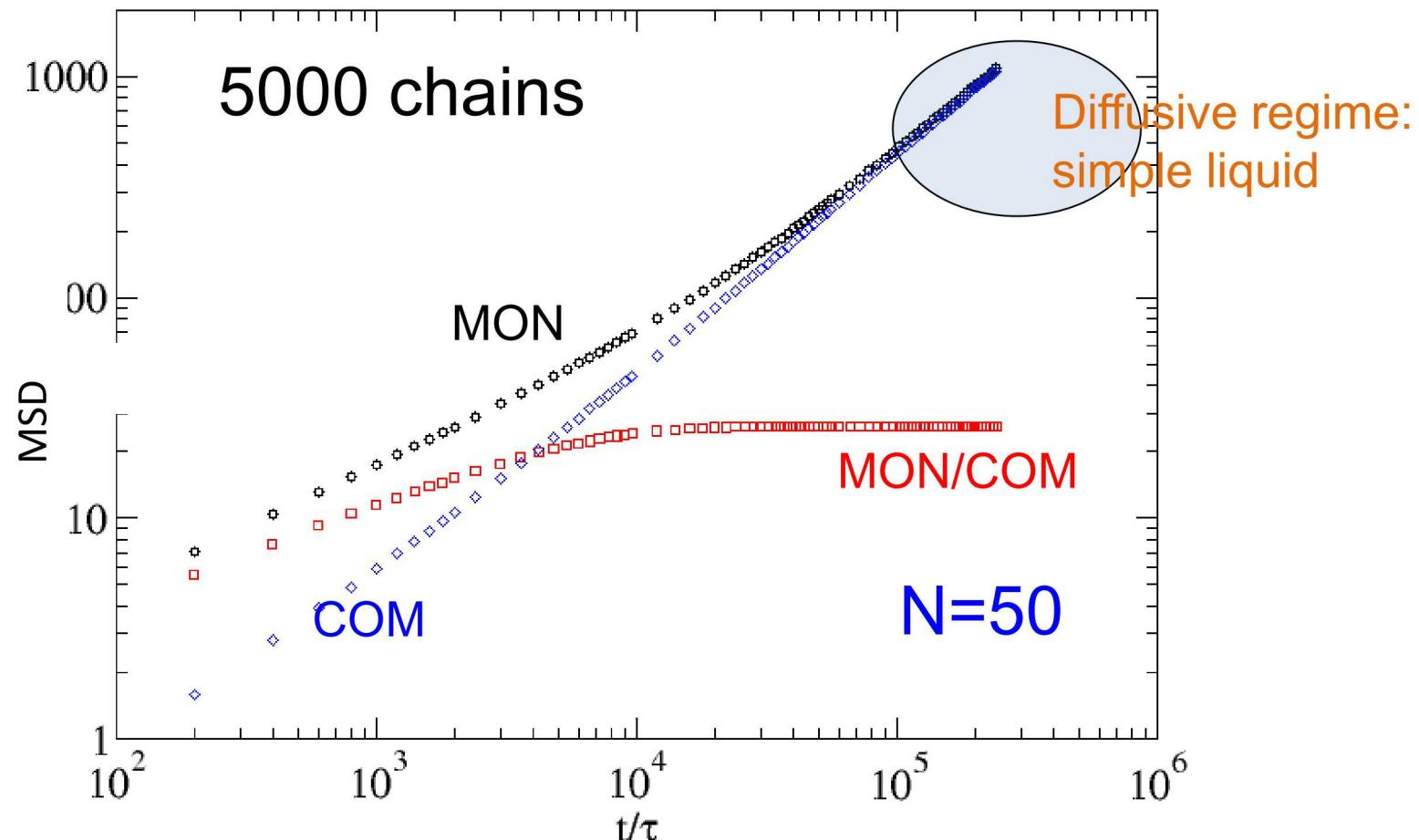


- Atomistic: All Atom



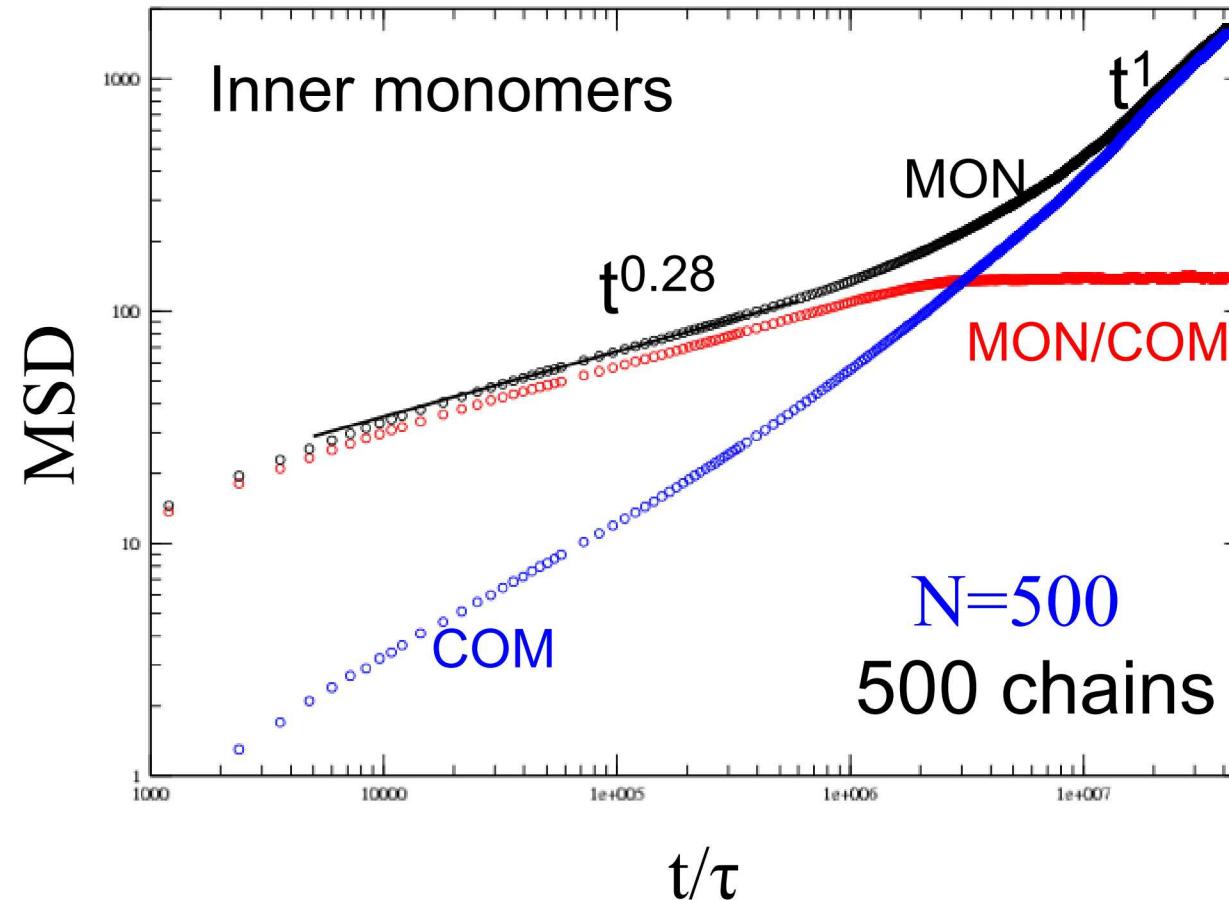
Multiple  $CH_2$   
combined into  
one bead

# Motion of Unentangled Polymer



- Once polymer move their own size, unentangled polymers move like simple liquids

# State of the Art: Motion of Entangled Polymer



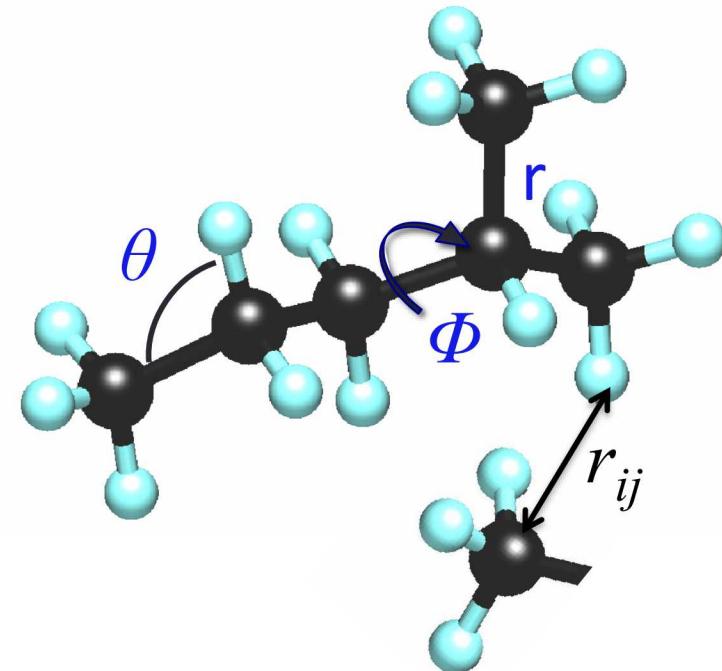
*2 million  
core hours*

*>10 billion  
time steps*

- $t^{1/4}$  reptation motion is clearly seen at intermediate times
- Second  $t^{1/2}$  region still unresolved

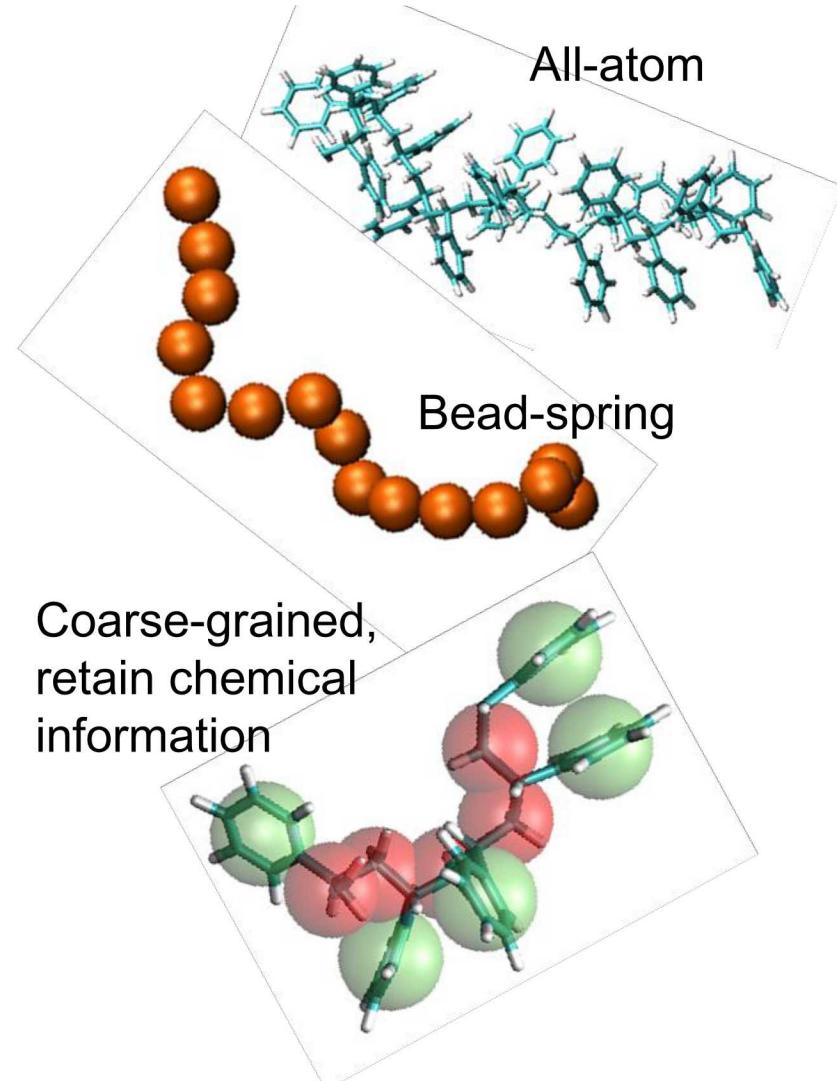
# Atomistic Simulations

- Bond, Angle, Dihedral
  - quantum chemistry
- Non-bonded van der Waals
  - empirically determined
  - Lennard-Jones 12-6, exp-6
- Coulomb interactions
  - partial/full charge
- Time step  $\sim 1.0\text{-}2.0$  femtoseconds  
 $1\text{ ns} = 10^6$  steps
- Presently limited to 100-1000 ns, 100's thousand – few million atoms



# Coarse-Graining of Polymers

- To reach larger length/time scales, new coarse graining methods are an active area of research



- Reduced number of degrees of freedom, simpler interaction potentials, reducing the overall computational effort
- Larger time steps (2-20x)
- Reduced effective bead friction due to lower energy barriers and/or a smoother energy landscape
- Back-mapping to fully atomistic model

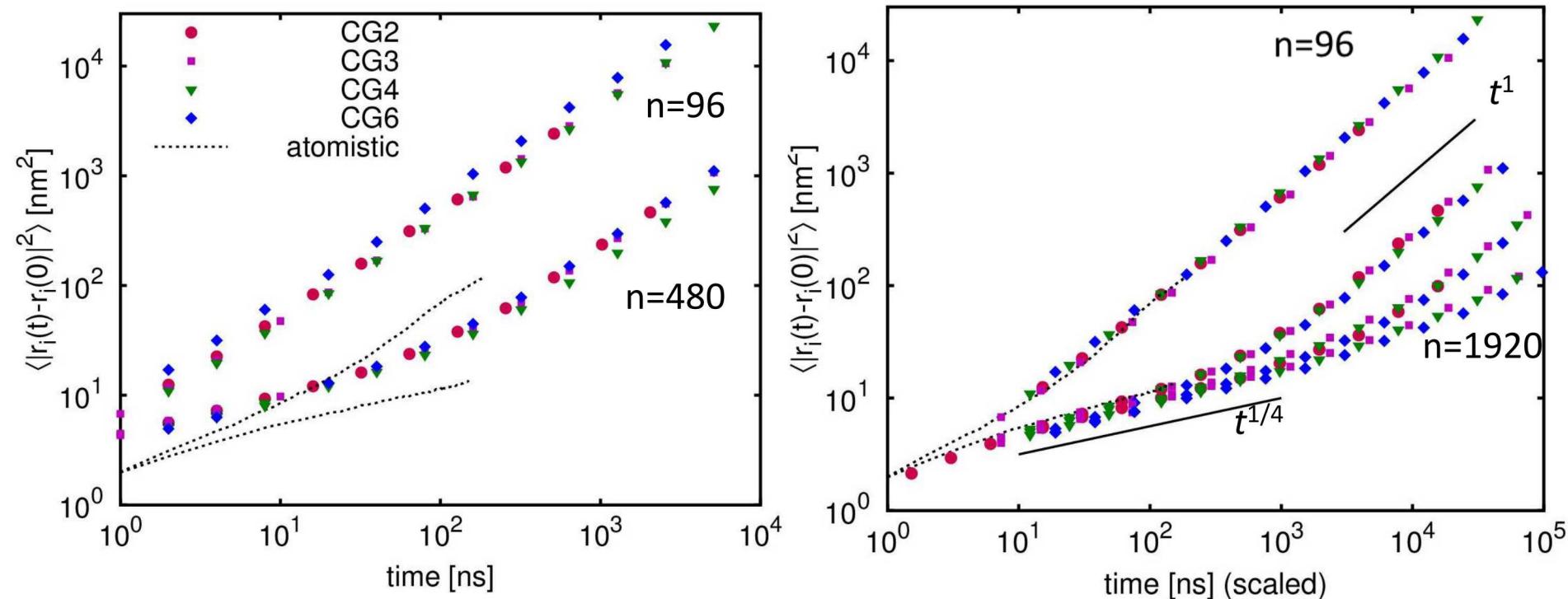
# Degree of Coarse Graining Polyethylene



$\text{C}_{96}\text{H}_{194}$  chain with increasing  
degree of coarse graining

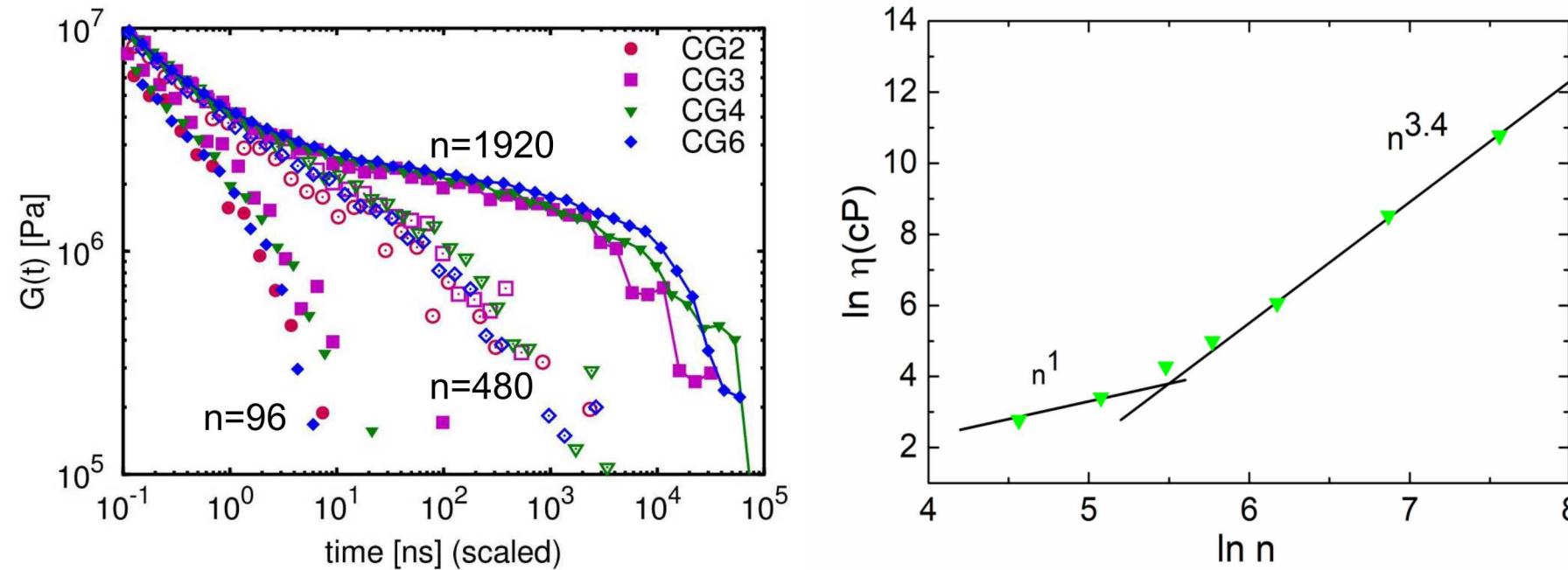
- Largest lengths scales of polymer dynamics are controlled by entanglements
- Shortest time and length scales required to resolve dynamic properties not obvious
- Probe the degree of coarse graining (CGing) required to simultaneously retain significant atomistic detail and access large length and time scales

# Mapping Dynamics: Coarse Grained to Atomistic



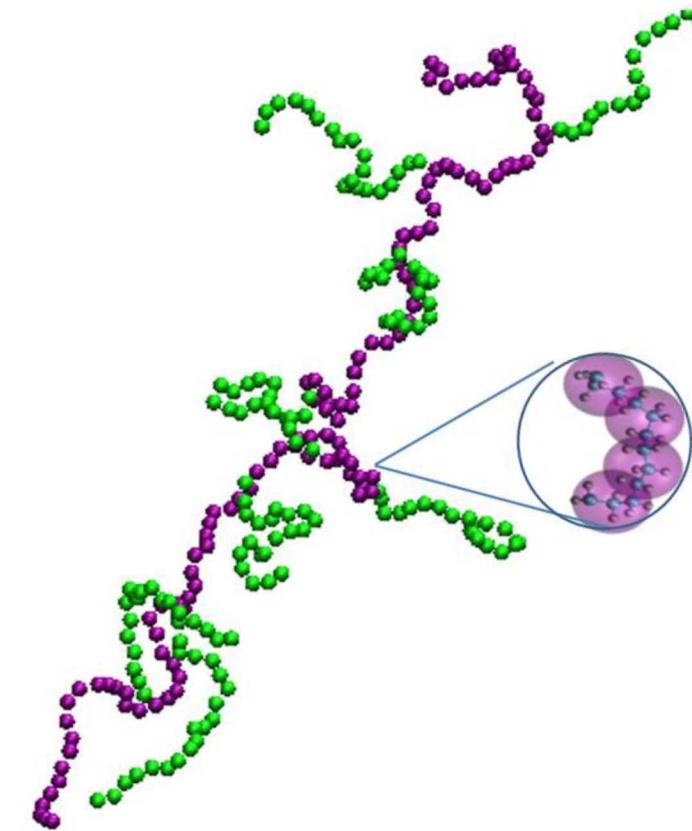
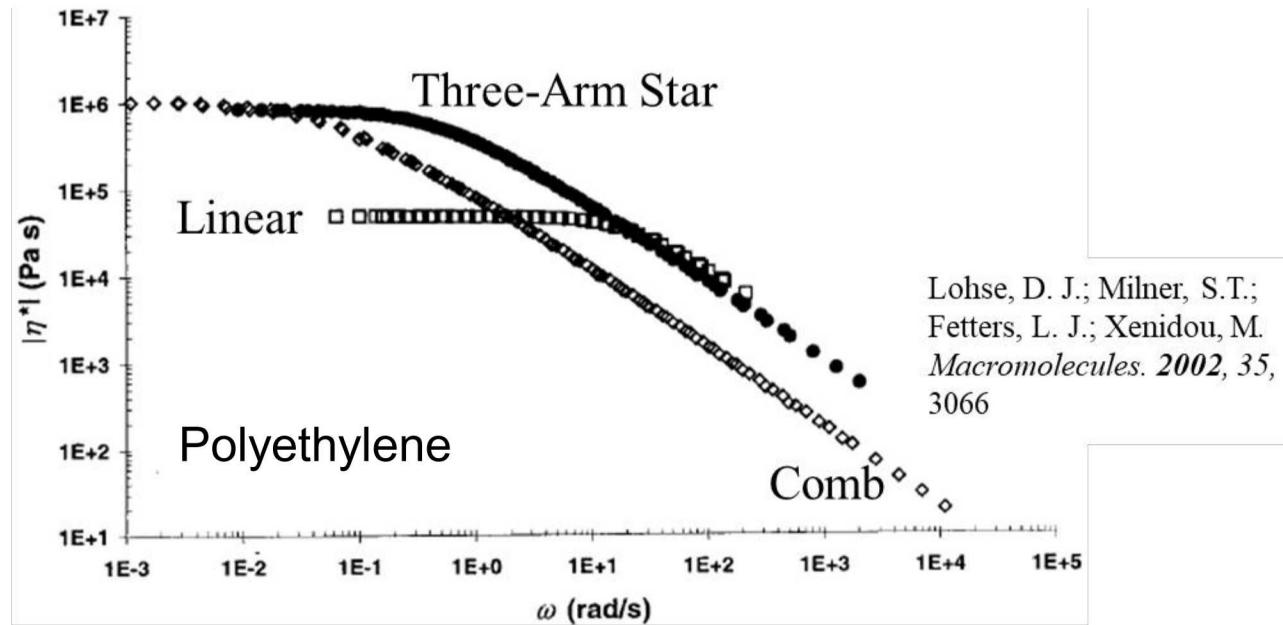
- Coarse graining reduces the number of degrees of freedom in a system, creating a smoother free-energy landscape
- Dynamics of Coarse Grained models 6-12 times faster
- Consistent scaling factor for  $n = 96 - 1920$

# Stress Relaxation and Viscosity



- Longer, more entangled chains form progressively more distinct plateau region
  - Plateau modulus in good agreement with experiment
- Viscosity versus shear rate show shear thinning at high shear rates, crossing over to shear independent regime
- Time and length scales not accessible by atomistic models

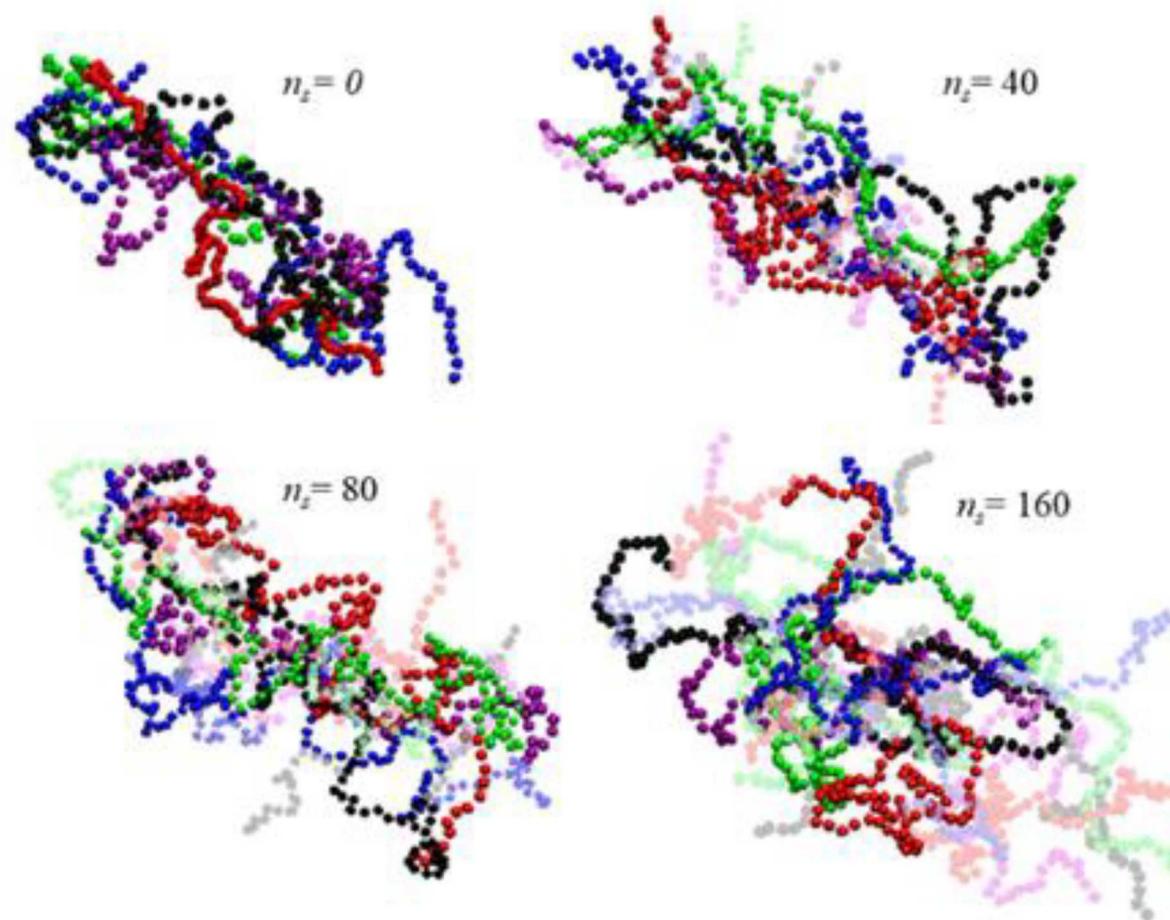
# Polymer Topology Effects on Rheology



- Branching changes the onset of shear thinning

# Motion of Comb Polymers

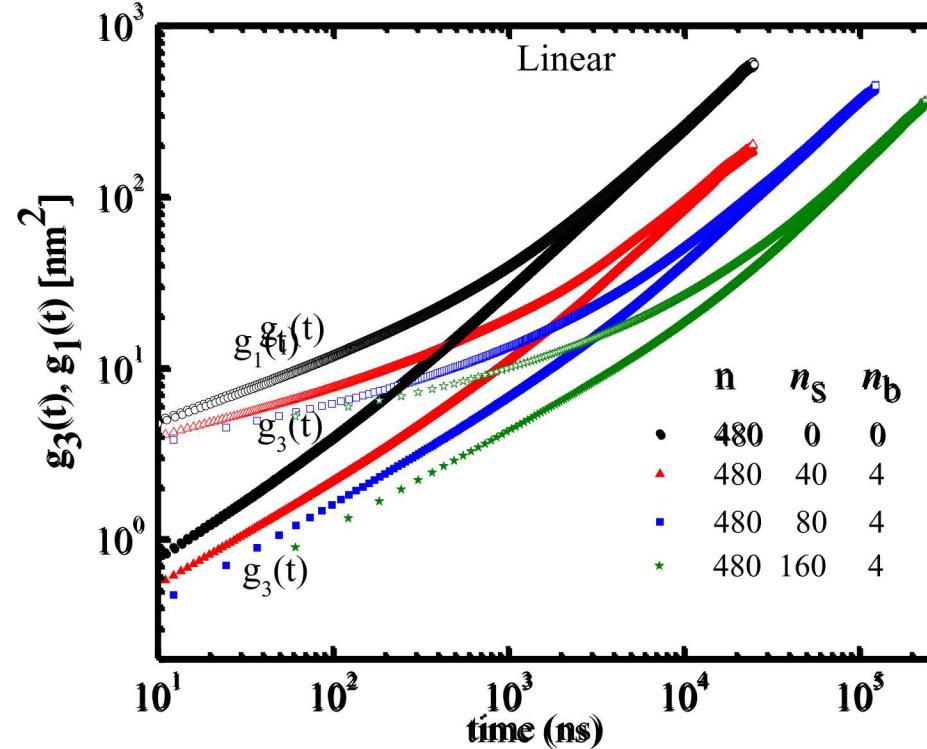
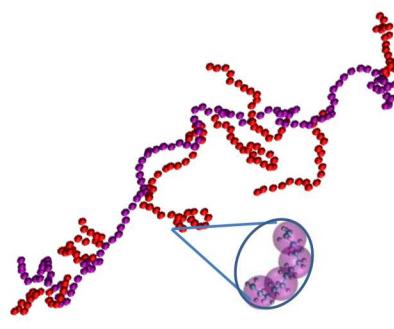
Backbone 120 CG  
Beads (480 Carbons)  
4 branches



5 Snapshot –  
62 ns apart

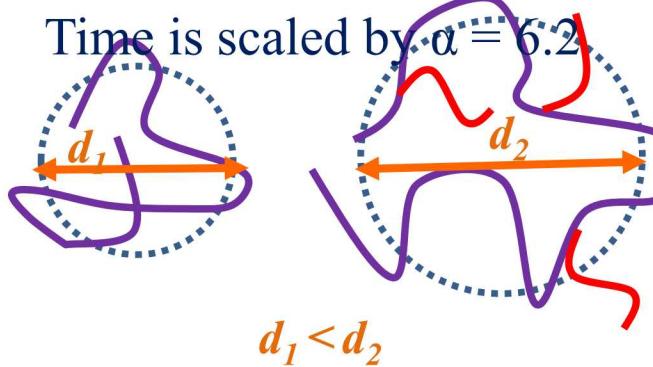
- Tube diameter Increases with Increasing branch length

# Effects of Branch Length on Chain Mobility



S. Wijesinghe et al., Macromolecules 51, 7621 (2018)

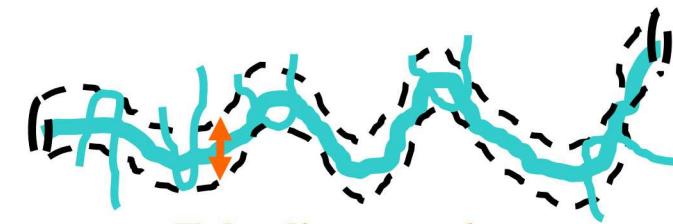
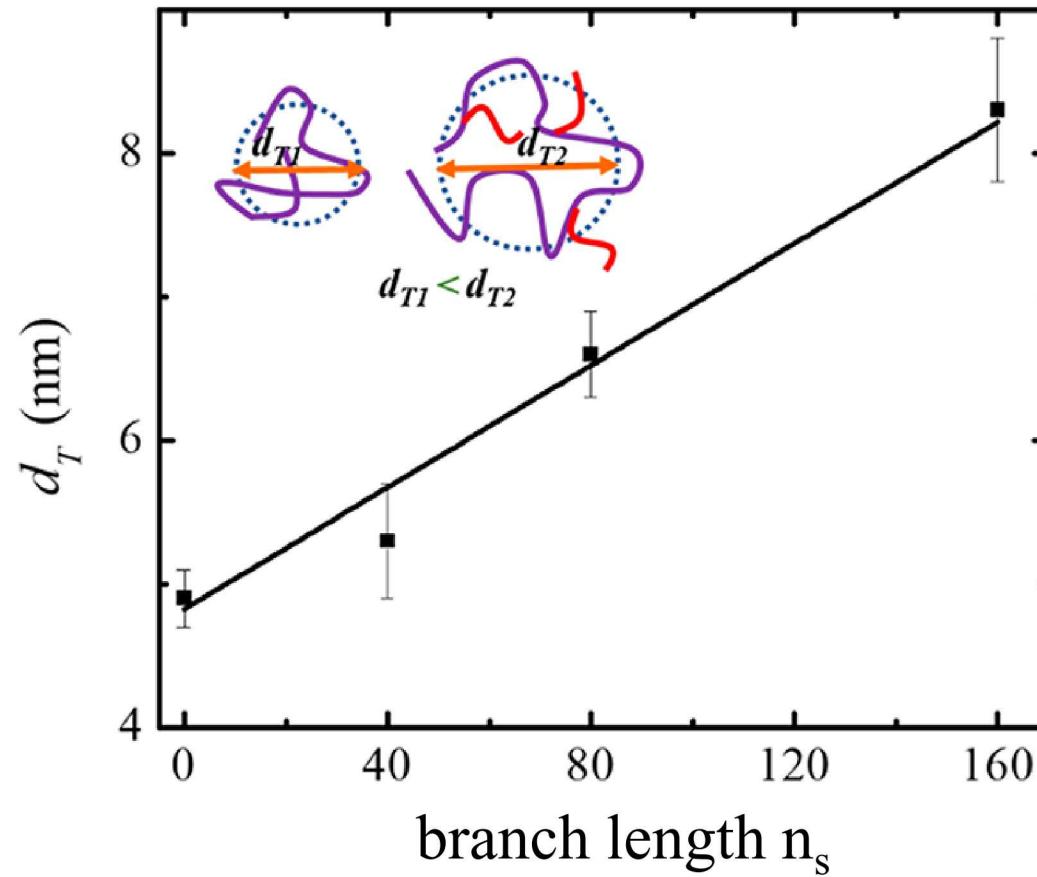
- Increasing branch length results in slower mobility of chains
- Motion of branching points similar to motion of inner monomers



*Cross-section of tube for linear and branched polymers*

# Effects of Branch Length on Tube Diameter

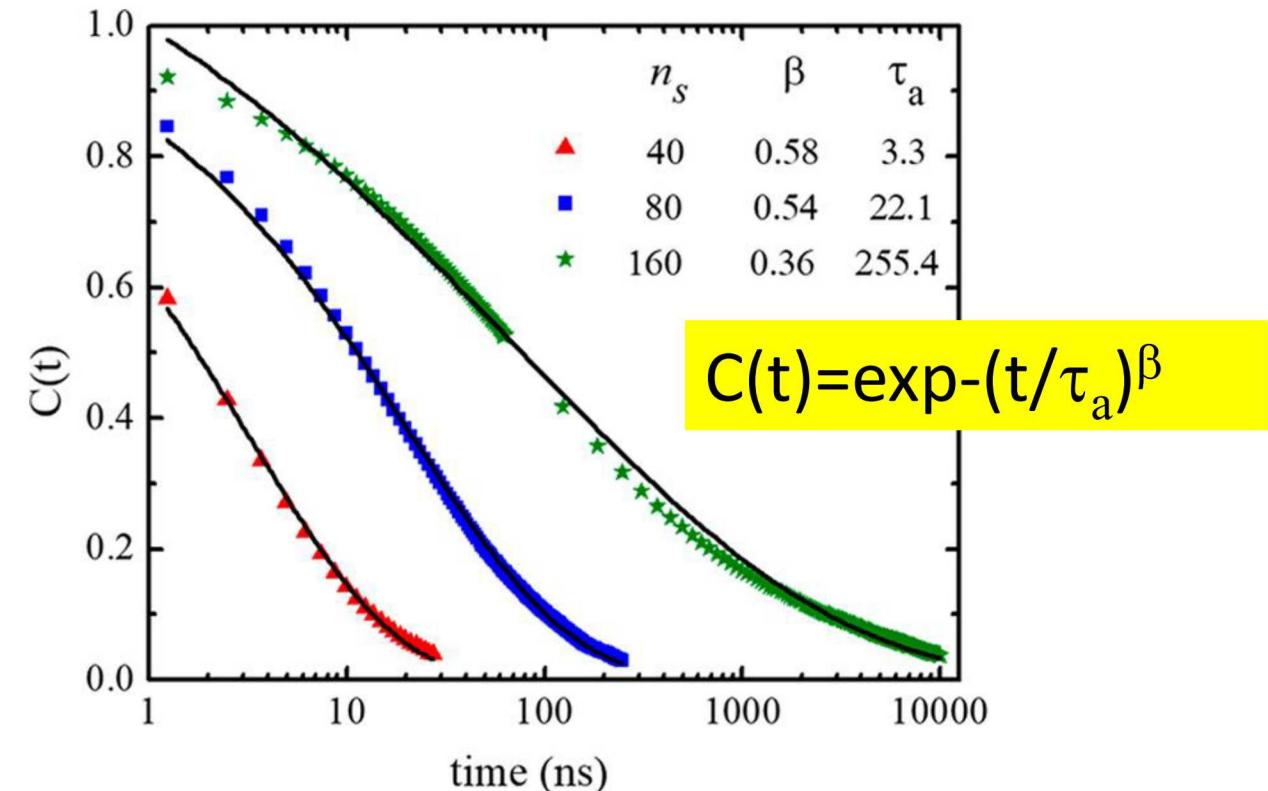
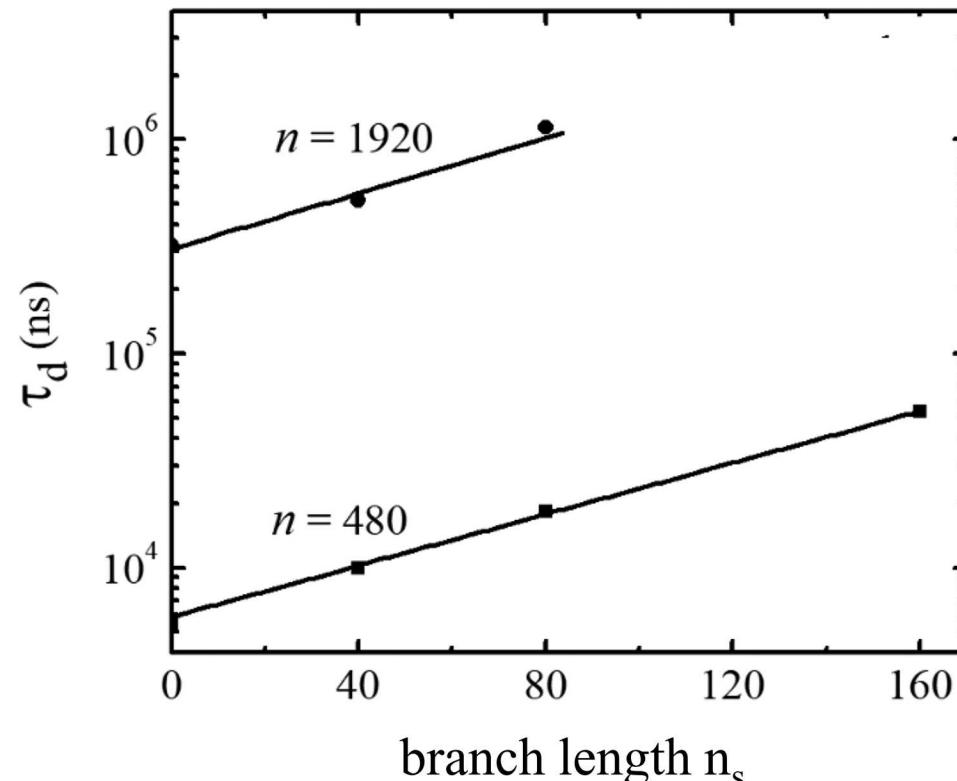
- Extract tube diameter  $d_T$  from crossover from early time  $t^{1/2}$  Rouse to  $t^{1/4}$  reptation regime



*Constrained polymer chain  
reptating in a tube*

Tube diameter increases  
linearly with branch length  $n_s$

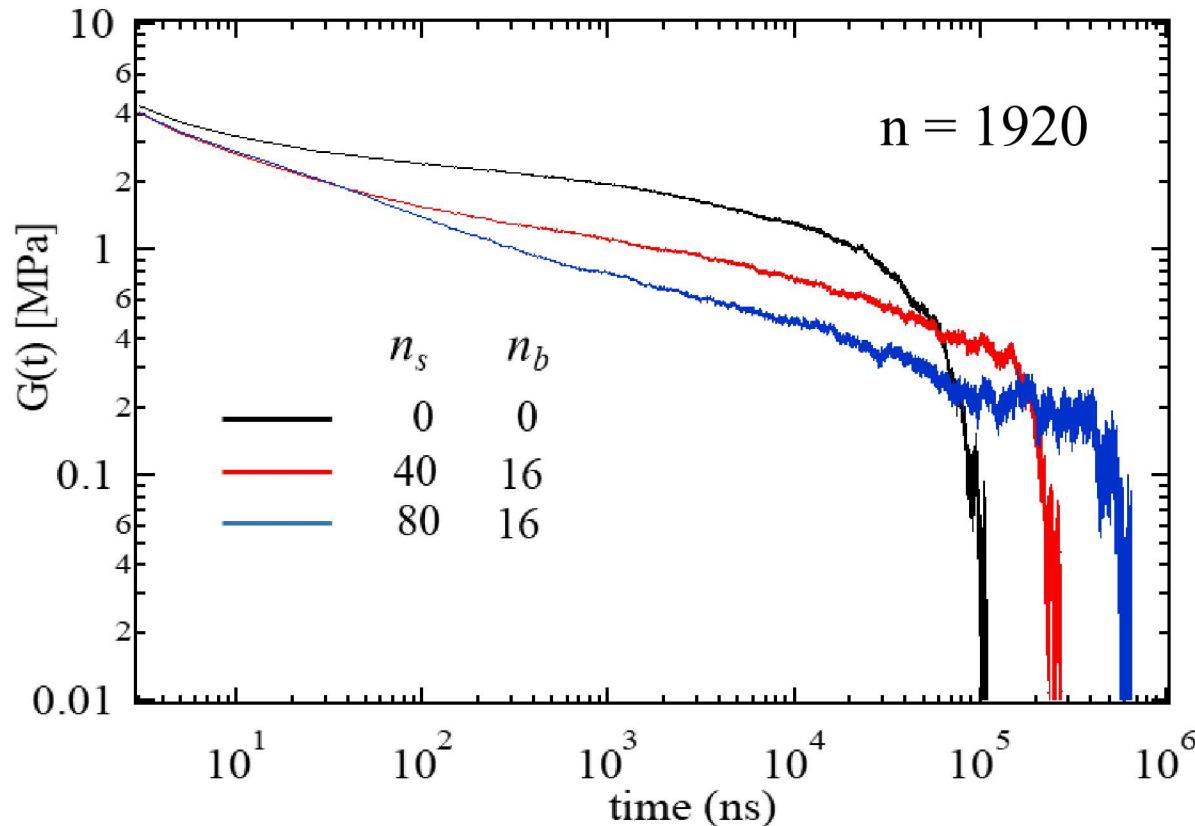
# Effects of Branch Length on Chain Relaxation



- Diffusive time increases exponentially with branch length

- End-to-end correlation of the branches has wide distribution of relaxation times

# Branching Effects on Stress Relaxation



□ Stress response function after a small perturbation  $G(t)$  can be expressed by the stress autocorrelation function:

$$G(t) = (V/k_B T) \langle \sigma_{\alpha\beta}(t) \sigma_{\alpha\beta}(0) \rangle$$

$\sigma_{\alpha\beta}$  off diagonal element of stress tensor

- Plateau modulus decreases with decreasing branch length  $n_s$  consistent with reduction in entanglement length and increase in tube diameter

# Summary/Outlook

- Atomistic Simulations ideally suited for phenomena on local scale
  - Present limitations 100-1000's nanoseconds, 10's nanometers
  - Exascale Computing will extend time and length scale significantly
- Simple Coarse-Grained models ideally suited for addressing general polymer phenomena, testing basic theoretical models
  - Disregards atomistic details
  - Can not quantitatively describe properties like structure, local dynamics
- Systematic coarse grained models can bridge the gap of time and length scales while retaining atomistic characteristics
  - Reduces number of degrees of freedom and increases fundamental time step
  - Captures the atomistic detail needed for correct dynamics from monomer to polymer scale

# Future Outlook – Exascale and Beyond

- Coarse-Grained Models:
  - Extend beyond focus on chain mobility
  - Stress Relaxation
  - Elongational and Shear Flow
  - Complex Architectures
  - Dispersity
  - Polymer Nanocomposites – Tethered Chains/Shape
- Atomistic Simulations
  - Hundreds ns/day
  - Extend times to 10's-100's  $\mu$ s
  - Multi-million atoms simulations

# Acknowledgements

## Collaborators:

- Anupriya Agrawal (Washington University, St. Louis)
- K. Michael Salerno (U S Army Research Lab)
- Ting Ge (Duke University)
- Dvora Perahia (Clemson University)
- Brandon Peters (University of Chicago)
- Sidath Wijesinghe (University of North Carolina)

## Funding:

- Center for Integrated Nanotechnologies (CINT)
- Laboratory Directed Research and Development (Sandia)