



Going up in time and length scales in modeling entangled polymers

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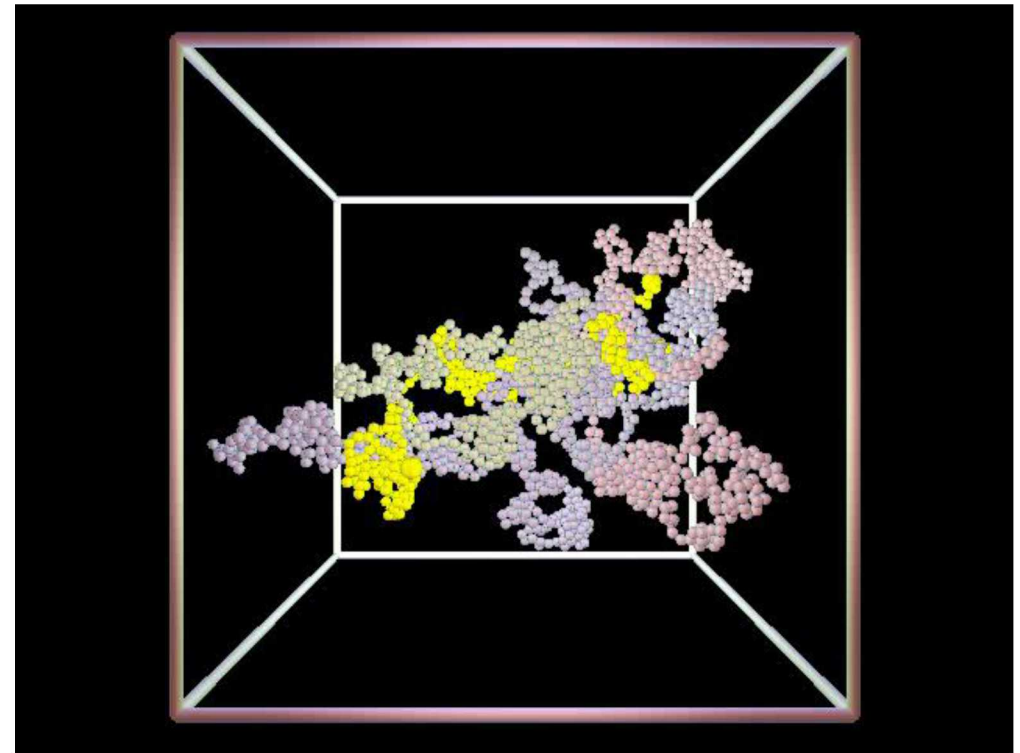
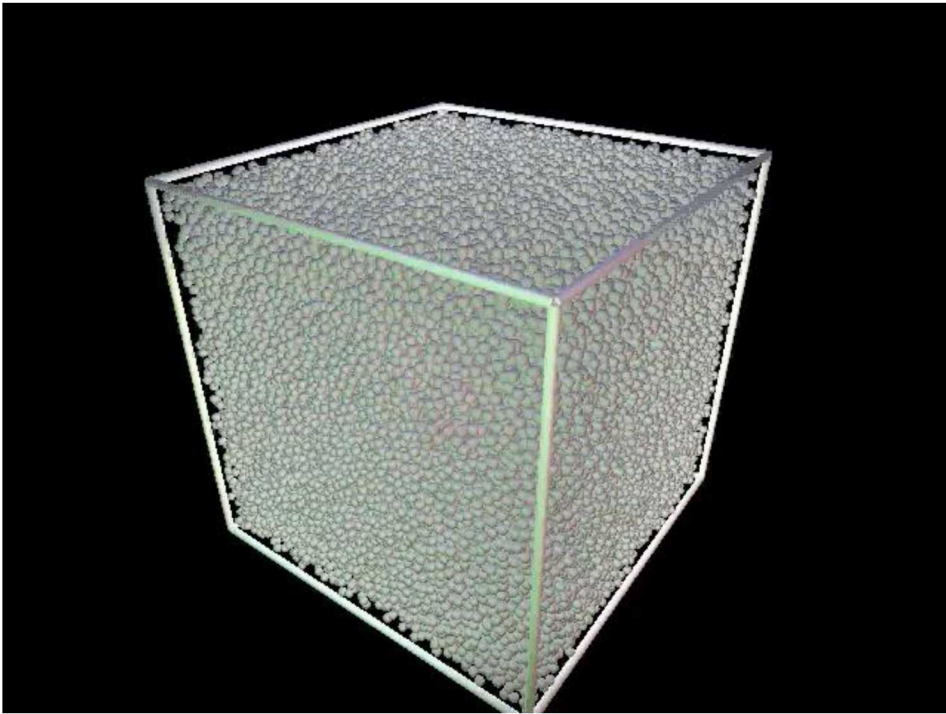
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Polymer Motion

- Polymers are simultaneously hard and soft
 - Unique Viscoelastic Behavior



- Motion of a polymer chain is subject to topological constraints

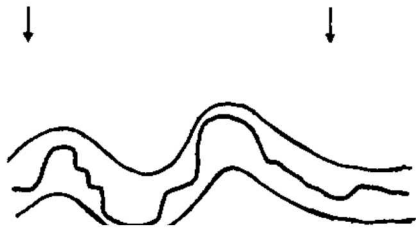
Polymer Motion in a Liquid



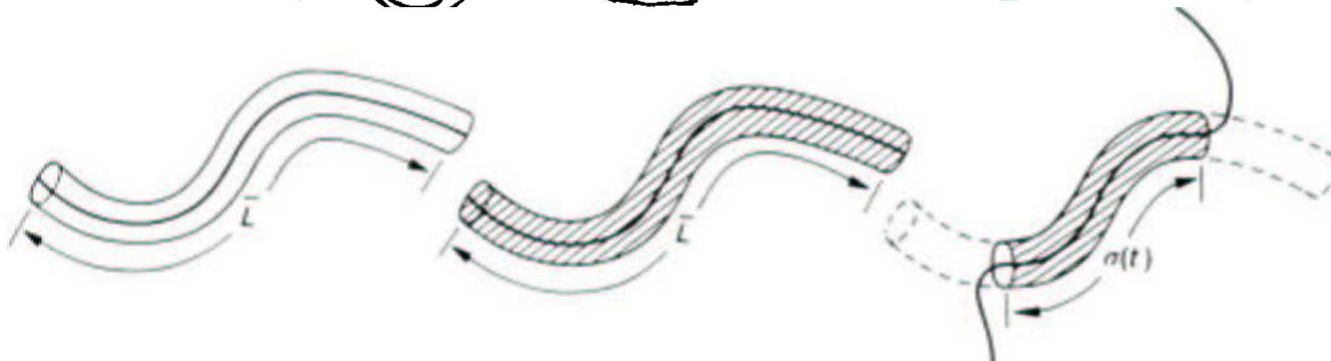
- Statistical Mechanics Approach



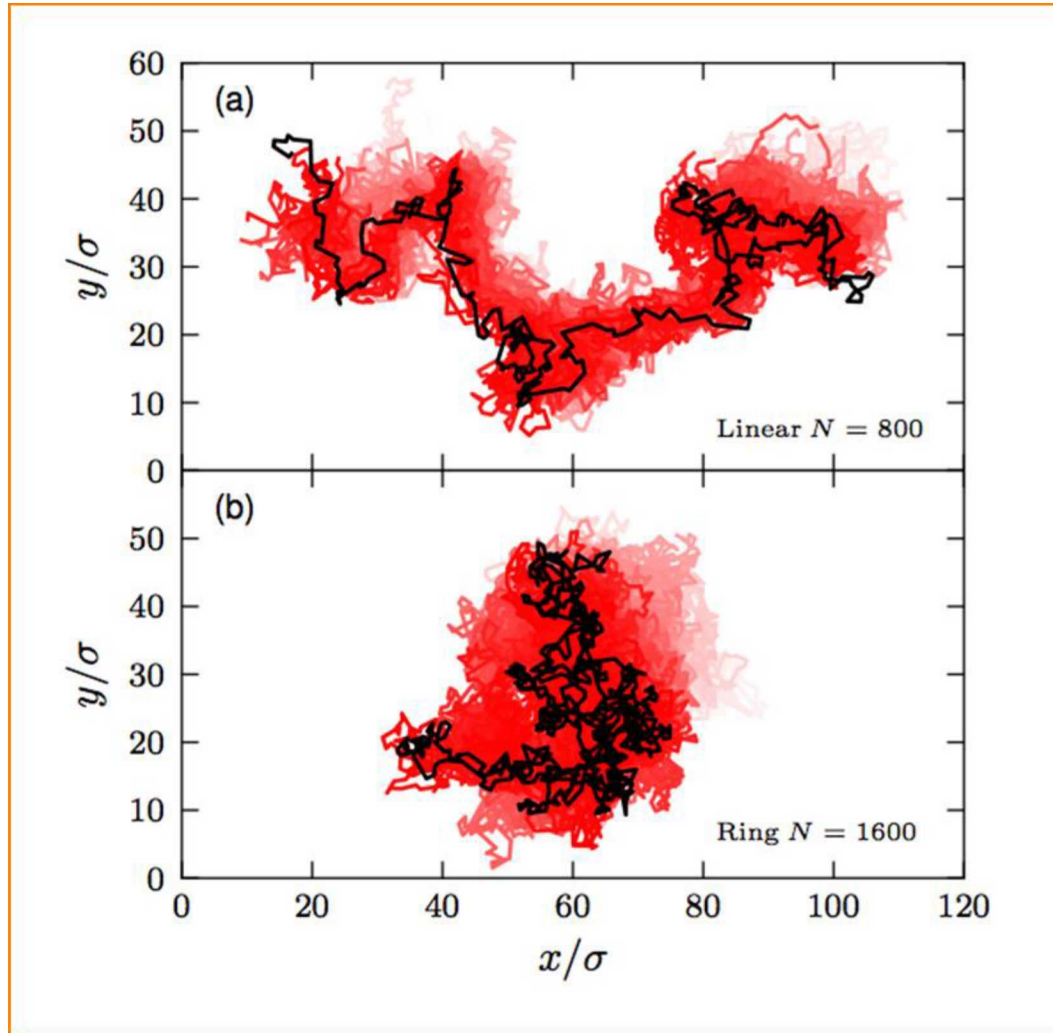
- Tube model (Edwards 1967)



- Reptation (de Gennes, 1971)



Polymer Topology Matters



- Linear chains move like a snake

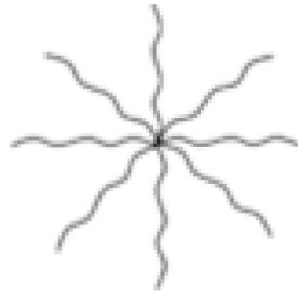
- Rings move by an amoeba-like motion

Polymer Architectures

Polymers not limited to linear chains



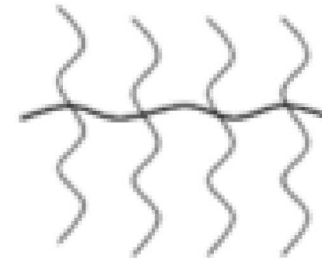
Block copolymer



Star polymer



Comb polymer



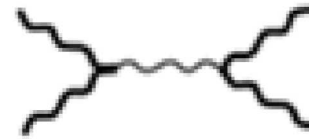
Brush polymer



AB_2 star



Palm-tree AB_n



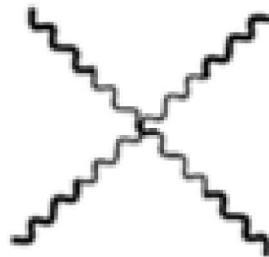
H-shaped B_2AB_2



Dumbbell (pom-pom)



Ring block



Star block AB_n



Coil-cycle-coil



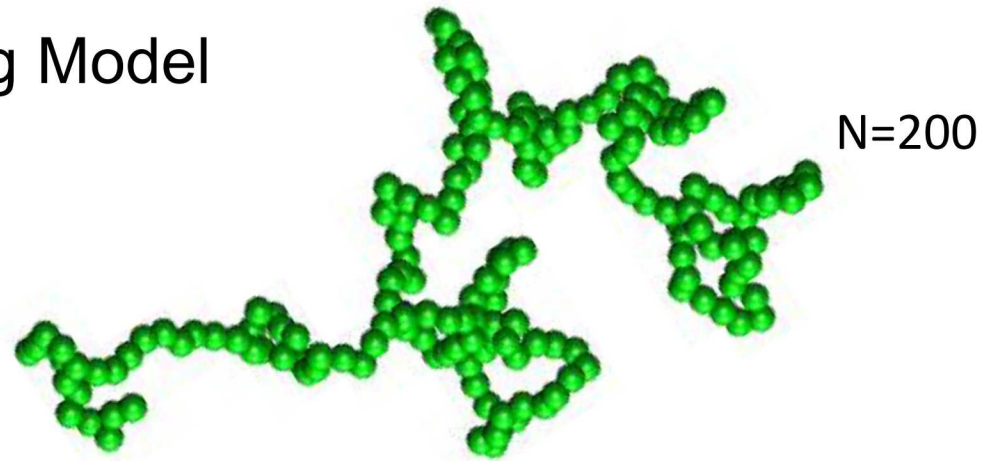
Star A_nB_n

Computational Challenges in Polymers

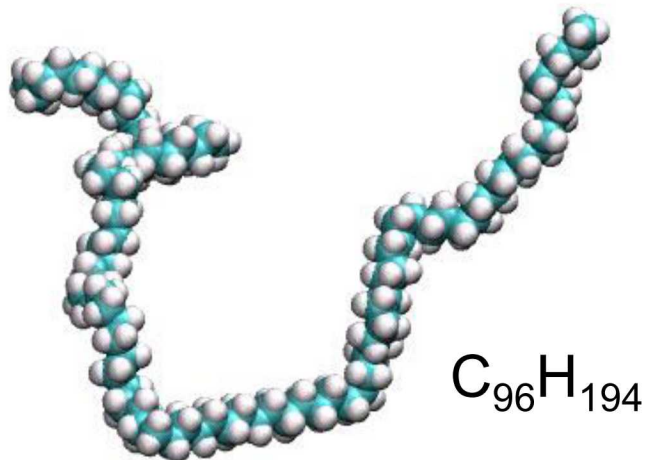
- Longest relaxation time $\tau \sim N^3$
- Chains are Gaussian coils – $R \sim N^{1/2}$
 - Number of chains must increase as $R^3 \sim N^{3/2}$ so polymer chains do not to see themselves through periodic boundary conditions
- Double chain length – cpu required increases by at least a factor of $2^{4.5} \sim 23$
 - 1-2 month simulation becomes 2-4 years
- Number of processors limited: ~ 400 -1000 particles/processor

Polymer Simulation Models

- Bead-Spring Model

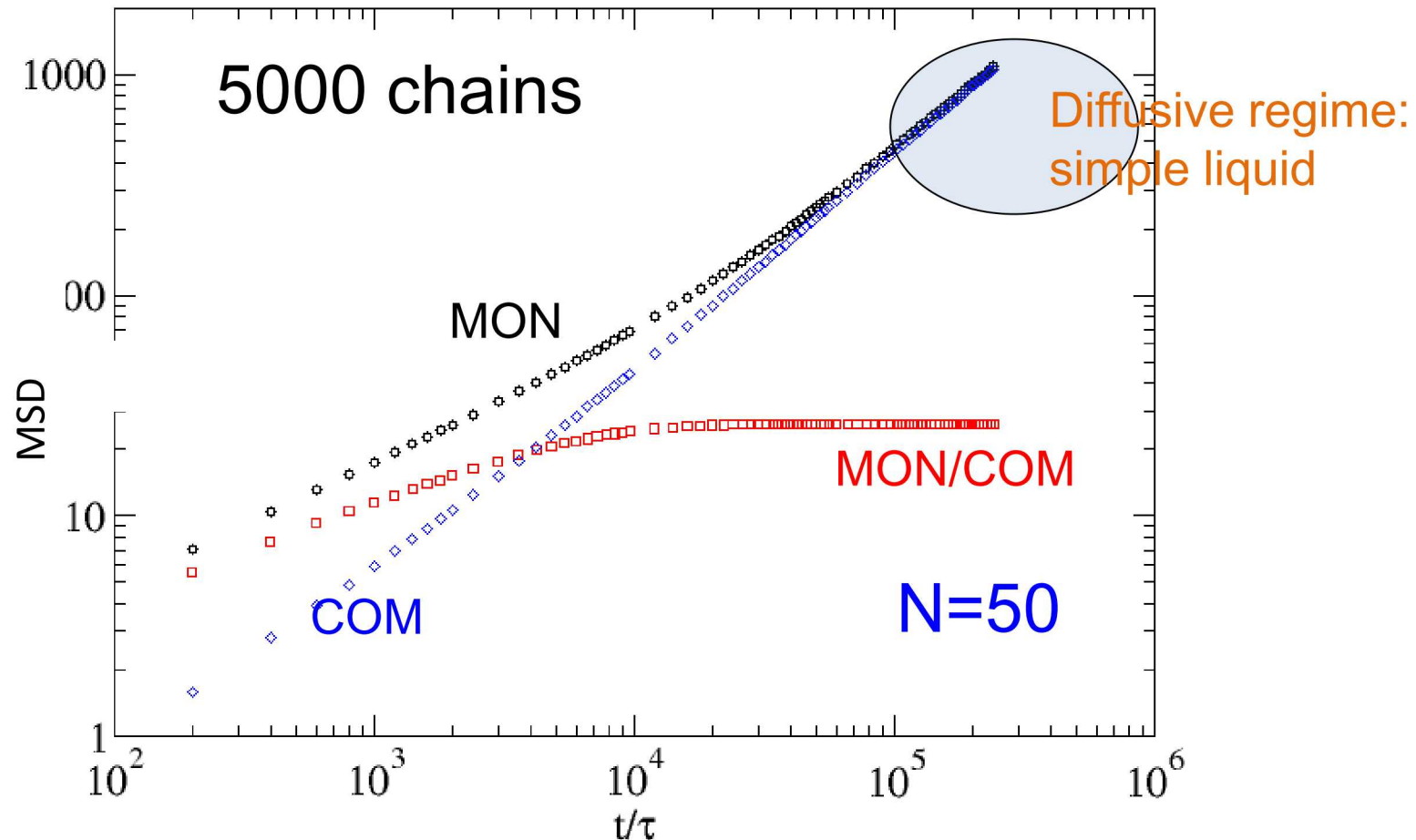


- Atomistic: All Atom



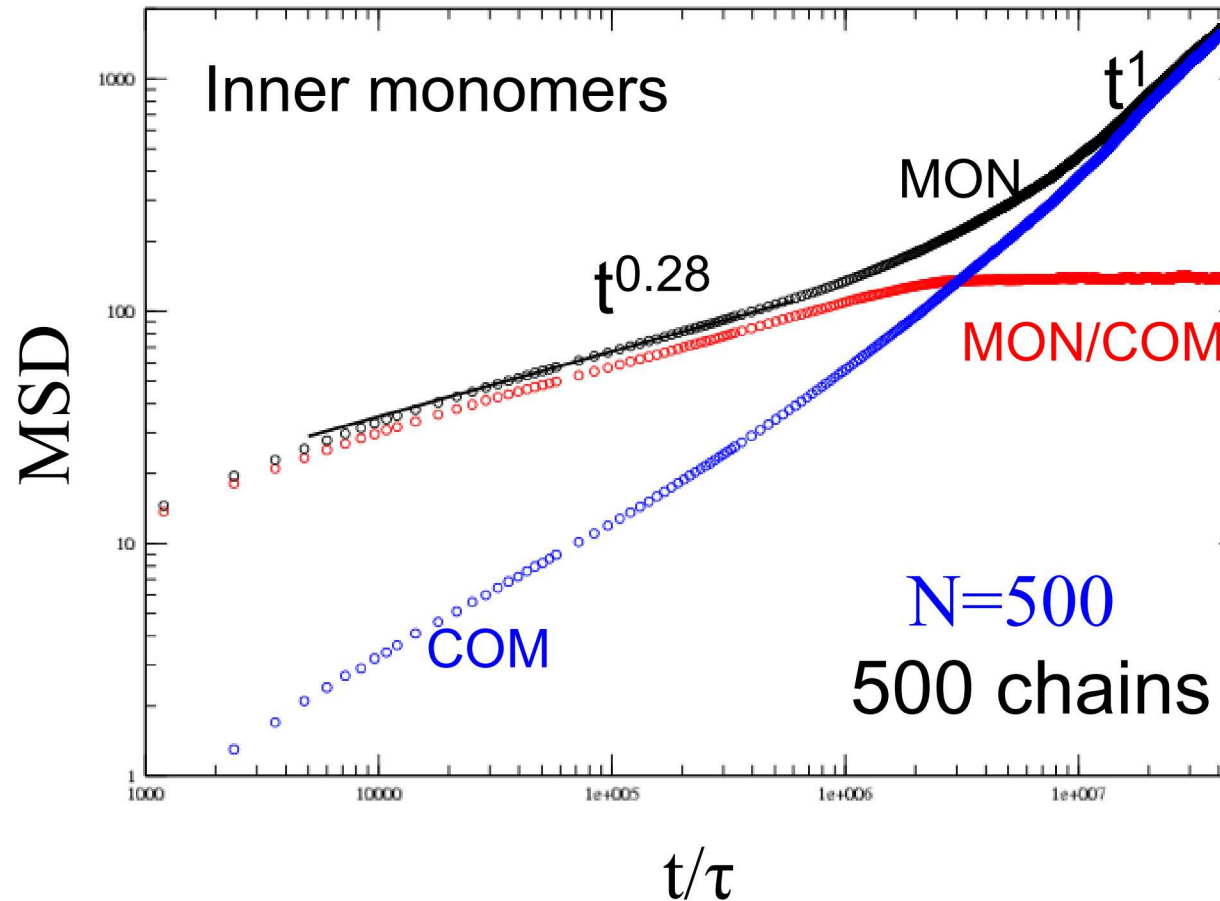
Multiple CH₂
combined into
one bead

Motion of Unentangled Polymer



- Once polymer move their own size, unentangled polymers move like simple liquids

State of the Art: Motion of Entangled Polymer



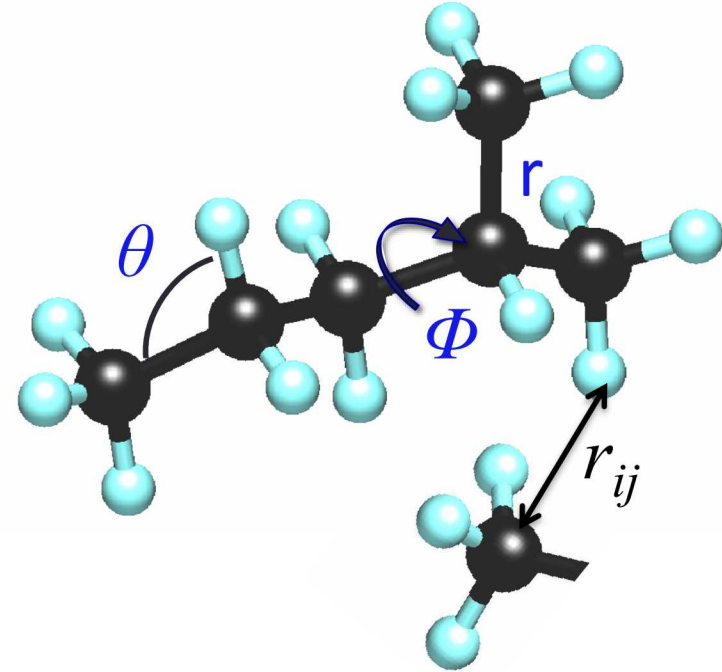
*2 million
core hours*

*>10 billion
time steps*

- $t^{1/4}$ reptation motion is clearly seen at intermediate times
- Second $t^{1/2}$ region still unresolved

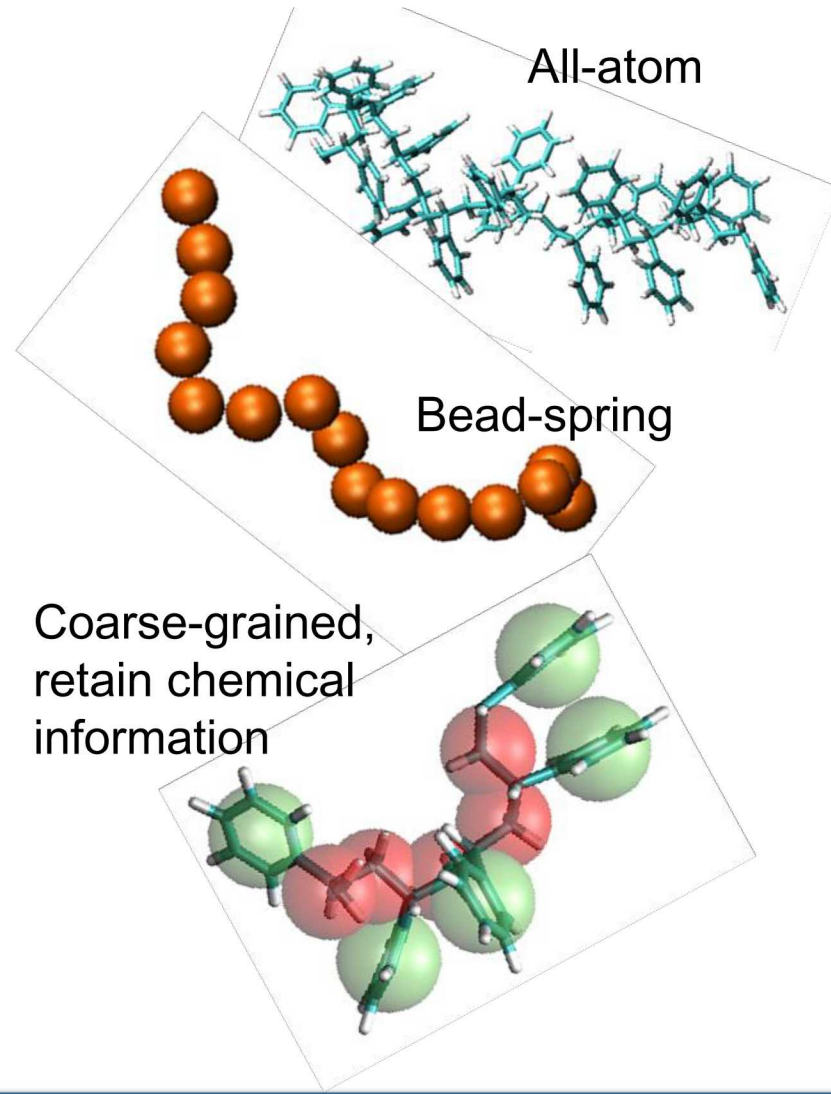
Atomistic Simulations

- Bond, Angle, Dihedral
 - quantum chemistry
 - Non-bonded van der Waals
 - empirically determined
 - Lennard-Jones 12-6, exp-6
 - Coulomb interactions
 - partial/full charge
-
- Time step $\sim 1.0\text{-}2.0$ femtoseconds
 $1\text{ ns} = 10^6$ steps
 - Presently limited to 100-1000 ns, 100's thousand – few million atoms



Coarse-Graining of Polymers

- To reach larger length/time scales, new coarse graining methods are an active area of research



- Reduced number of degrees of freedom, simpler interaction potentials, reducing the overall computational effort
- Larger time steps (2-20x)
- Reduced effective bead friction due to lower energy barriers and/or a smoother energy landscape
- Back-mapping to fully atomistic model

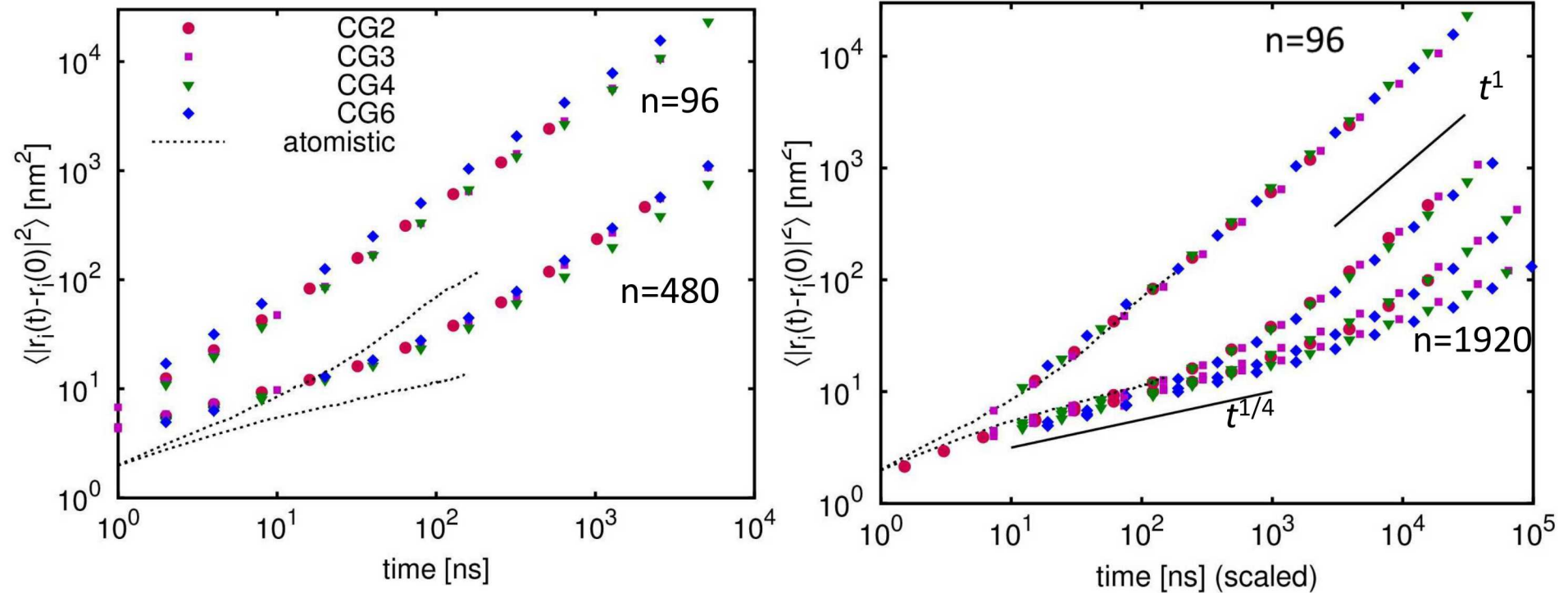
Degree of Coarse Graining Polyethylene



$C_{96}H_{194}$ chain with increasing
degree of coarse graining

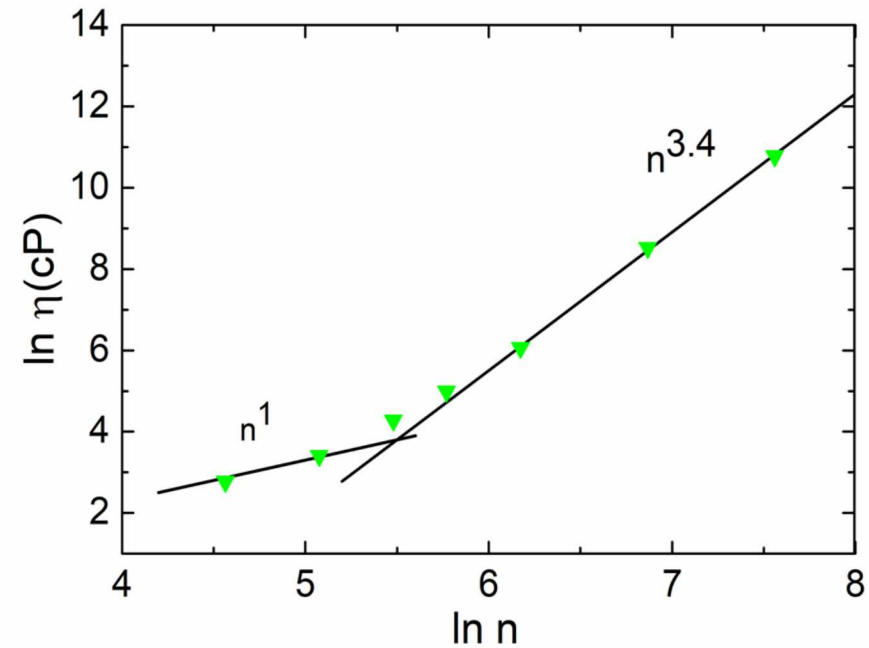
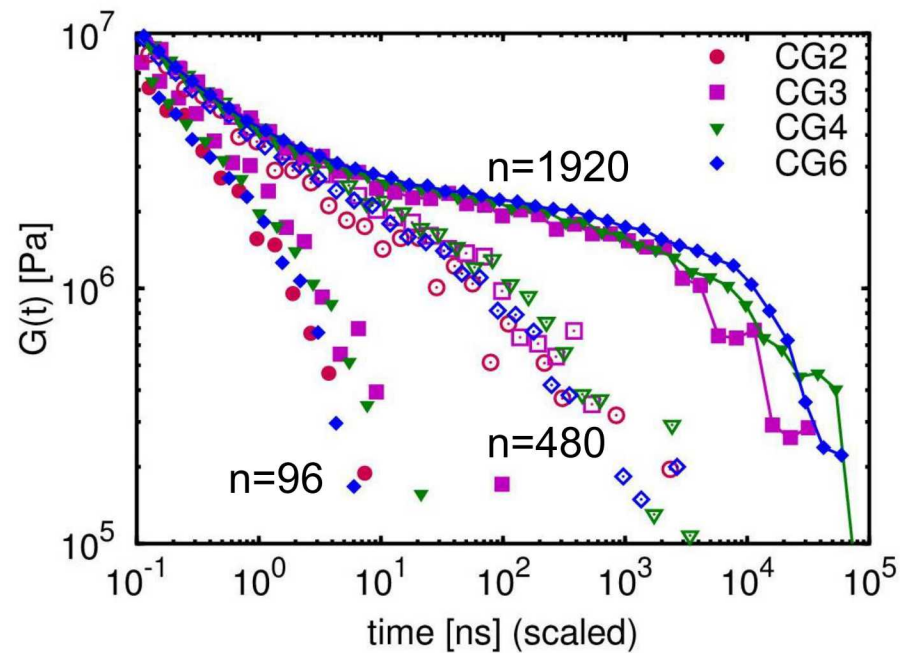
- Largest length scales of polymer dynamics are controlled by entanglements
- Shortest time and length scales required to resolve dynamic properties not obvious
- **Probe the degree of coarse graining (CGing) required to simultaneously retain significant atomistic detail and access large length and time scales**

Mapping Dynamics: Coarse Grained to Atomistic



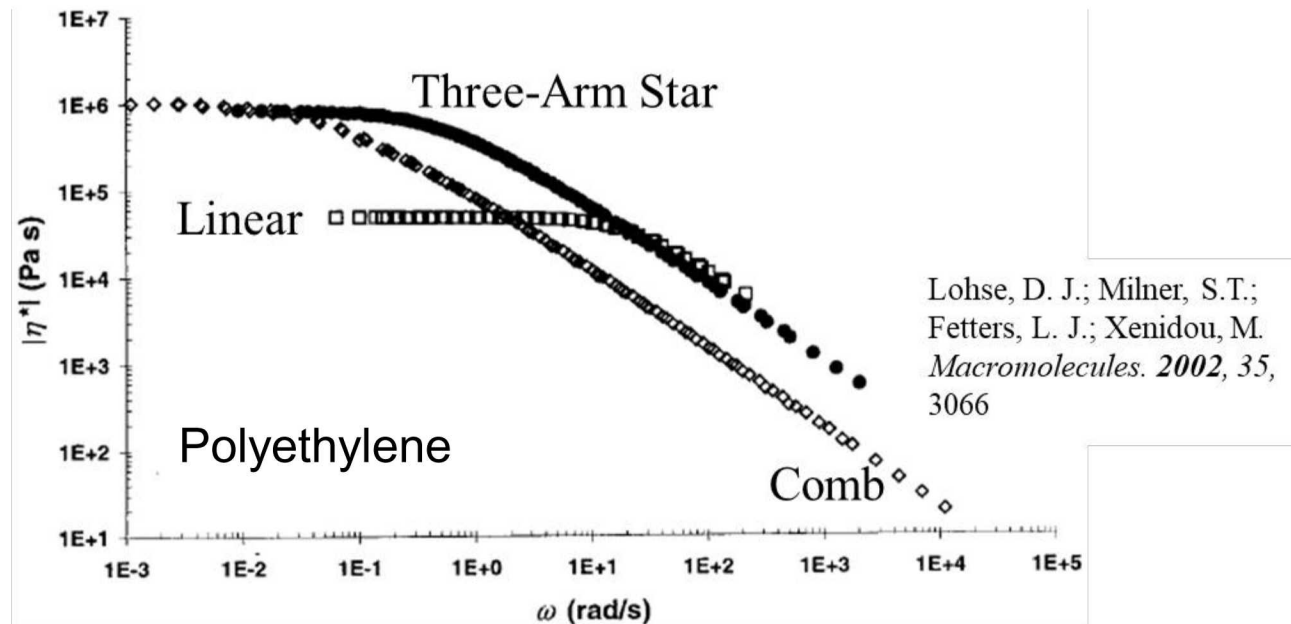
- Coarse graining reduces the number of degrees of freedom in a system, creating a smoother free-energy landscape
- Dynamics of Coarse Grained models 6-12 times faster
- Consistent scaling factor for $n = 96 - 1920$

Stress Relaxation and Viscosity

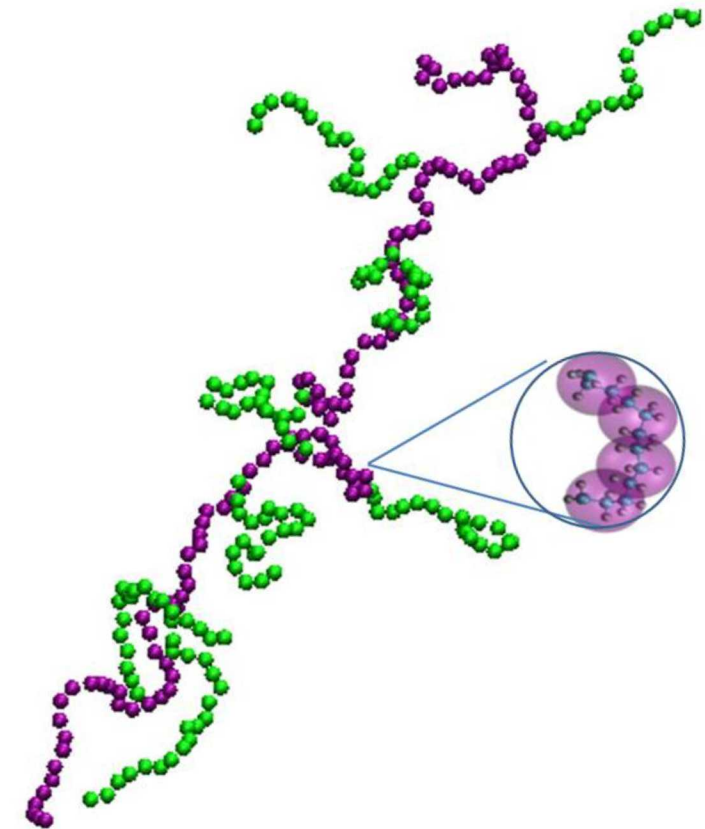


- Longer, more entangled chains form progressively more distinct plateau region
 - Plateau modulus in good agreement with experiment
- Viscosity versus shear rate show shear thinning at high shear rates, crossing over to shear independent regime
- Time and length scales not accessible by atomistic models

Polymer Topology Effects on Rheology

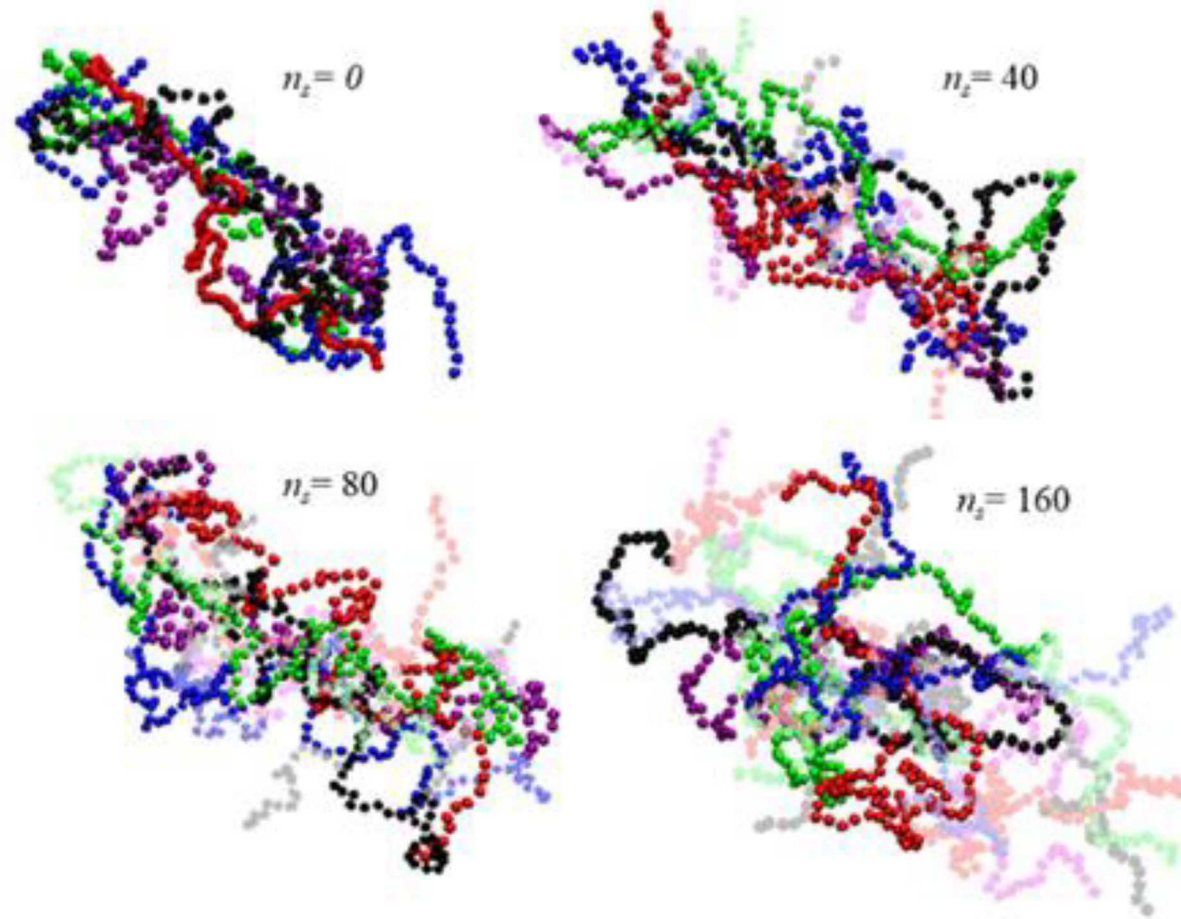


- Branching changes the onset of shear thinning



Coarse grained PE chain
with: $\lambda = 4$

Motion of Comb Polymers

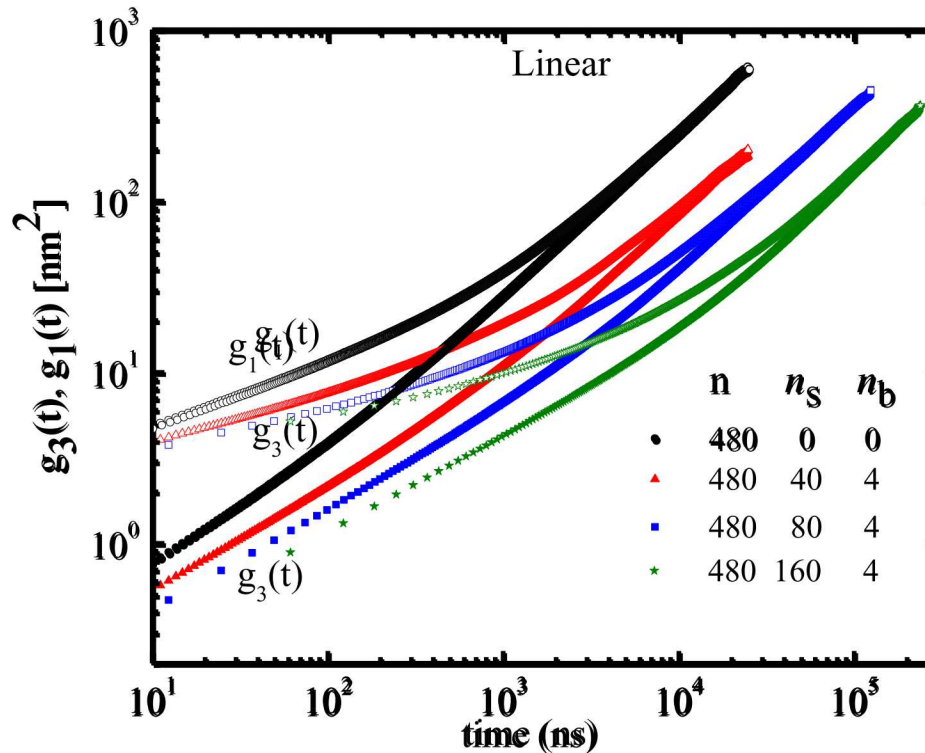
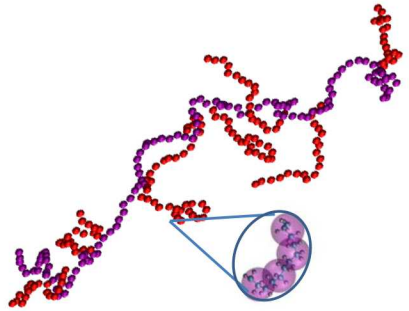


Backbone 120 CG
Beads (480 Carbons)
4 branches

5 Snapshot –
62 ns apart

- Tube diameter Increases with Increasing branch length

Effects of Branch Length on Chain Mobility

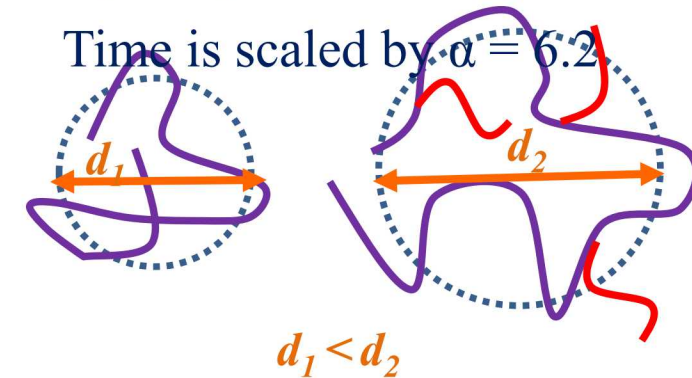


S. Wijesinghe et al., Macromolecules 51, 7621 (2018)

- Increasing branch length results in slower mobility of chains
- Motion of branching points similar to motion of inner monomers

$g_1(t)$ - mean squared displacement (MSD) of the center of mass of the chain
 Tube diameter d
 $g_3(t)$ - MSD of Center of Mass reptating in a tube

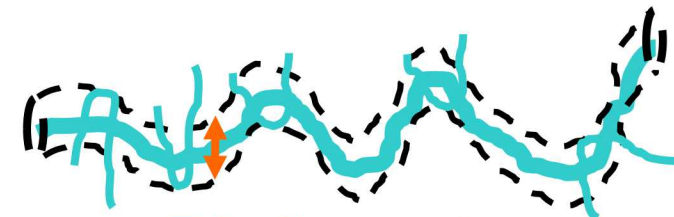
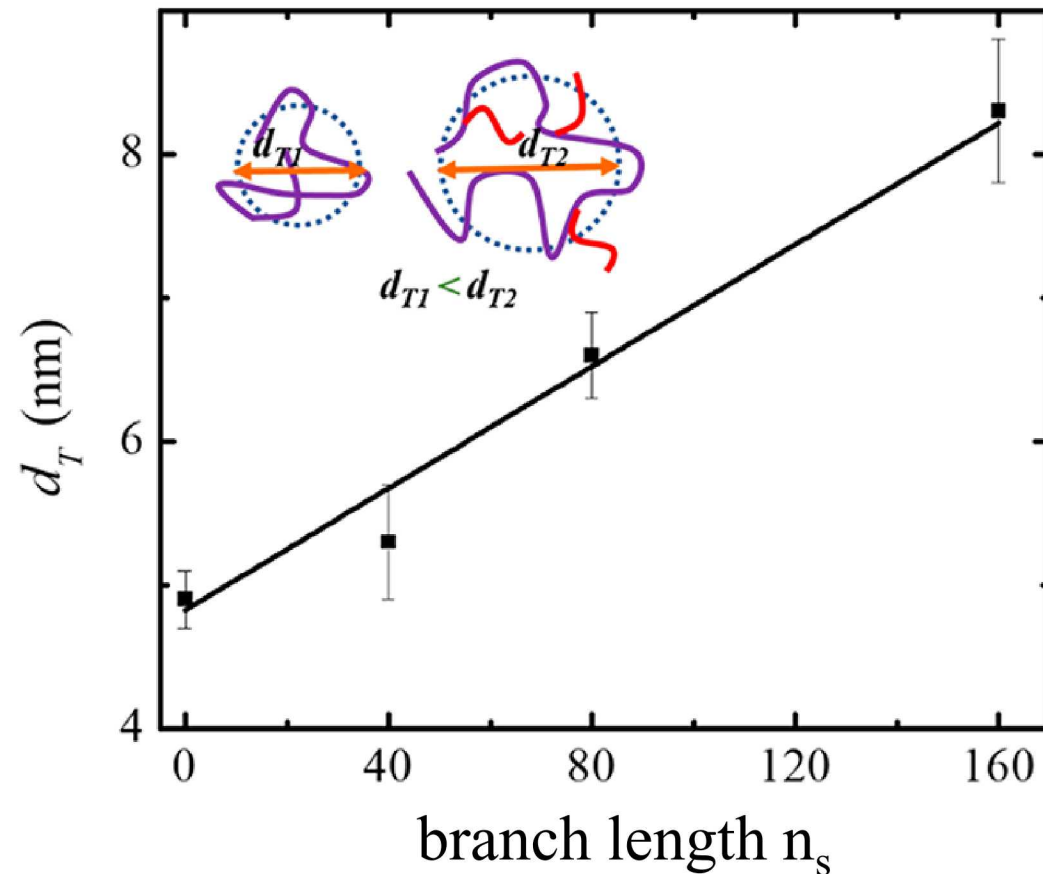
Time is scaled by $\alpha = 6.2$



$d_1 < d_2$
 Cross-section of tube for linear and branched polymers

Effects of Branch Length on Tube Diameter

- Extract tube diameter d_T from crossover from early time $t^{1/2}$ Rouse to $t^{1/4}$ reptation regime

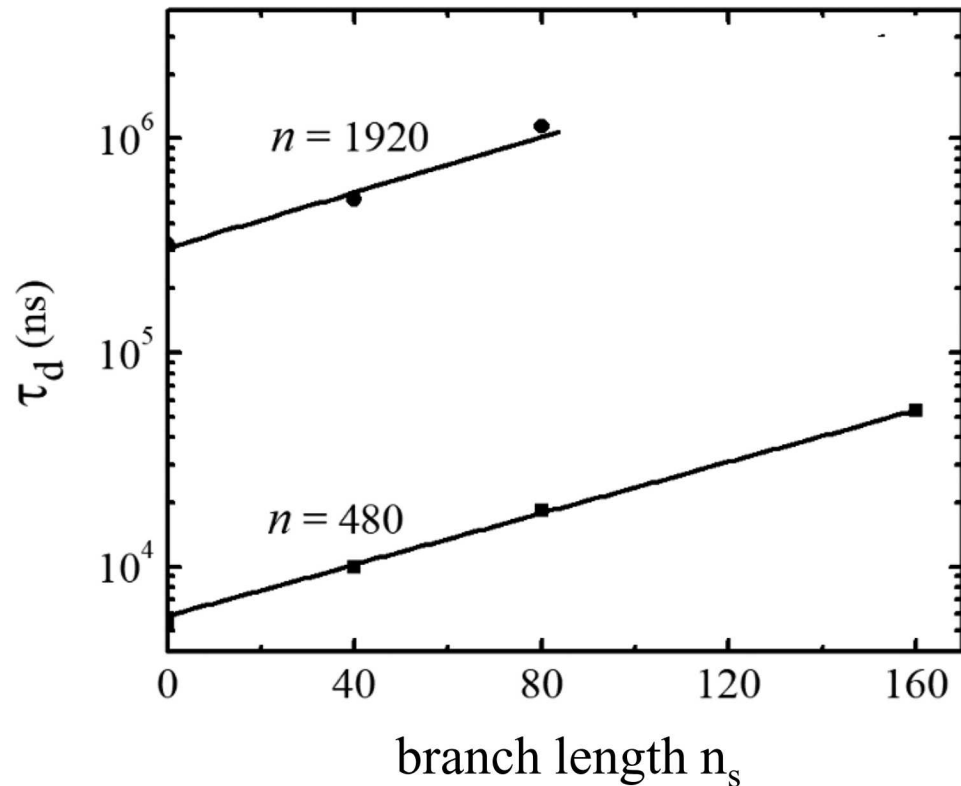


Tube diameter d

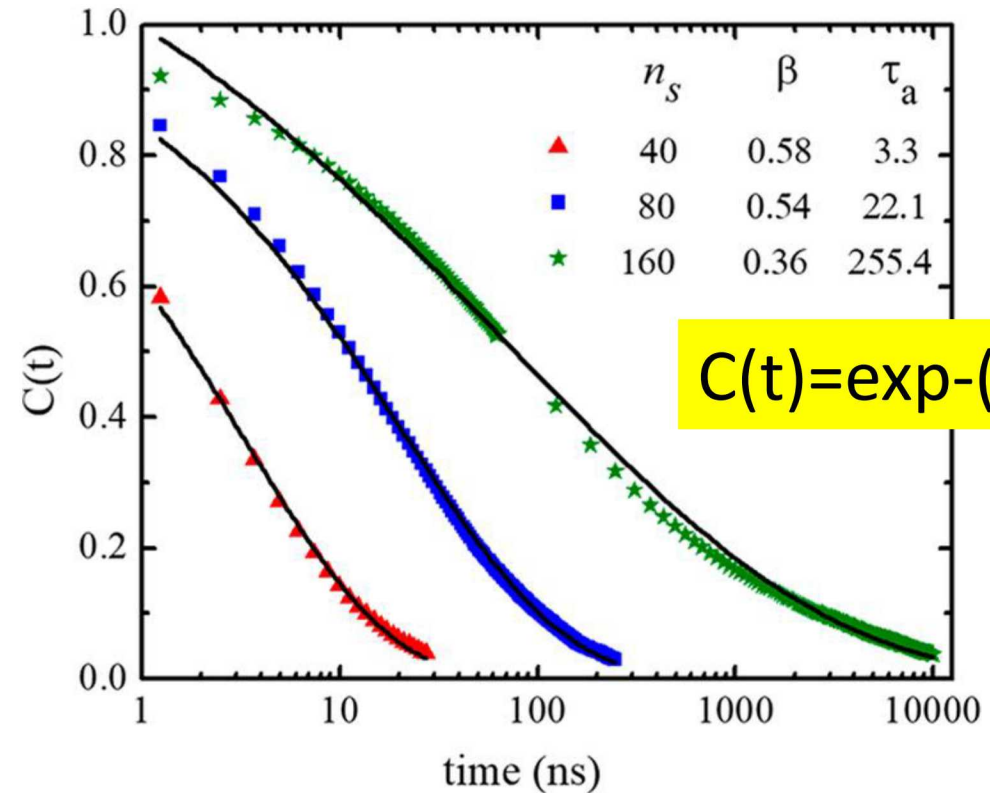
*Constrained polymer chain
reptating in a tube*

Tube diameter increases
linearly with branch length n_s

Effects of Branch Length on Chain Relaxation

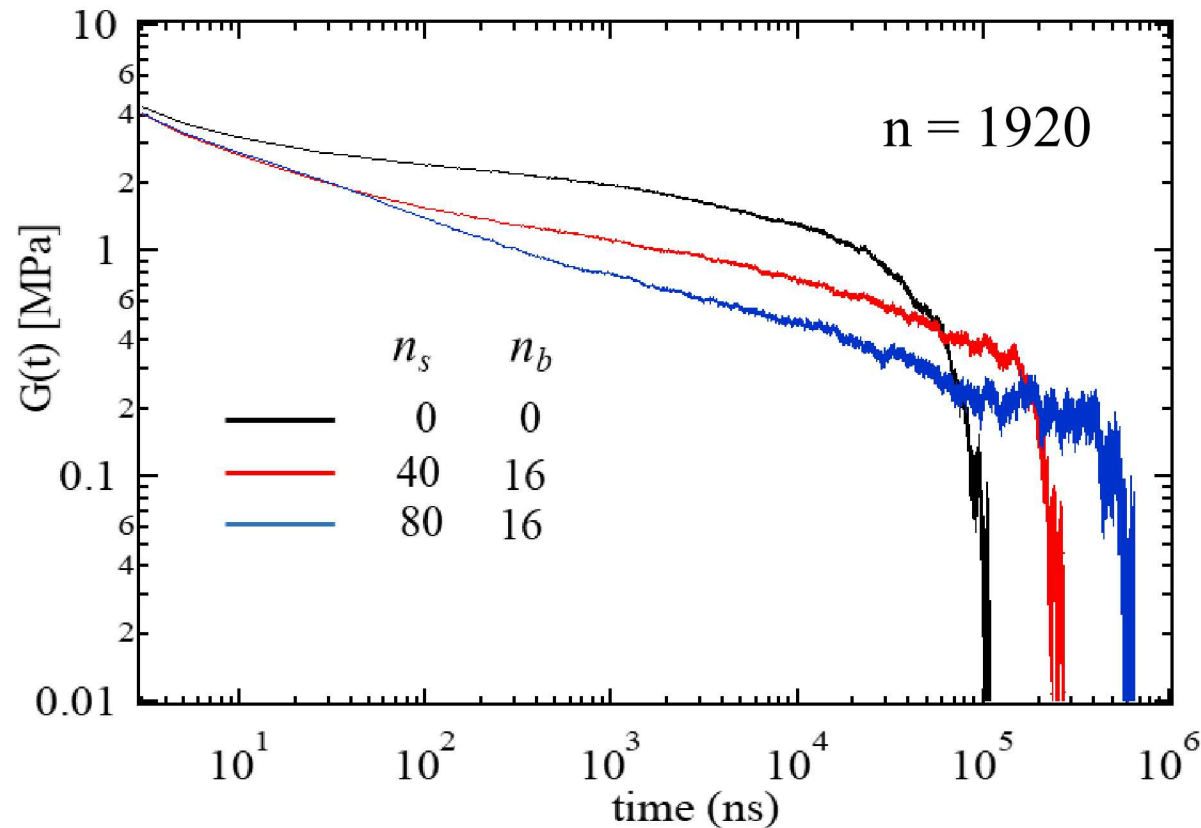


- Diffusive time increases exponentially with branch length



- End-to-end correlation of the branches has wide distribution of relaxation times

Branching Effects on Stress Relaxation



□ Stress response function after a small perturbation $G(t)$ can be expressed by the stress autocorrelation function:

$$G(t) = (V/k_B T) \langle \sigma_{\alpha\beta}(t) \sigma_{\alpha\beta}(0) \rangle$$

$\sigma_{\alpha\beta}$ off diagonal element of stress tensor

- Plateau modulus decreases with decreasing branch length n_s consistent with reduction in entanglement length and increase in tube diameter

Summary/Outlook

- Atomistic Simulations ideally suited for phenomena on local scale
 - Present limitations 100-1000's nanoseconds, 10's nanometers
 - Exascale Computing will extend time and length scale significantly
- Simple Coarse-Grained models ideally suited for addressing general polymer phenomena, testing basic theoretical models
 - Disregards atomistic details
 - Can not quantitatively describe properties like structure, local dynamics
- Systematic coarse grained models can bridge the gap of time and length scales while retaining atomistic characteristics
 - Reduces number of degrees of freedom and increases fundamental time step
 - Captures the atomistic detail needed for correct dynamics from monomer to polymer scale

Future Outlook – Exascale and Beyond

- Coarse-Grained Models:
 - Extend beyond focus on chain mobility
 - Stress Relaxation
 - Elongational and Shear Flow
 - Complex Architectures
 - Dispersity
 - Polymer Nanocomposites – Tethered Chains/Shape
- Atomistic Simulations
 - Hundreds ns/day
 - Extend times to 10's-100's μ s
 - Multi-million atoms simulations

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