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# Molecular Simulation of Geochemical Reactions in Subcritical Fractures



PRESENTED BY

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## Geochemical Reactions in Subcritical Fracture

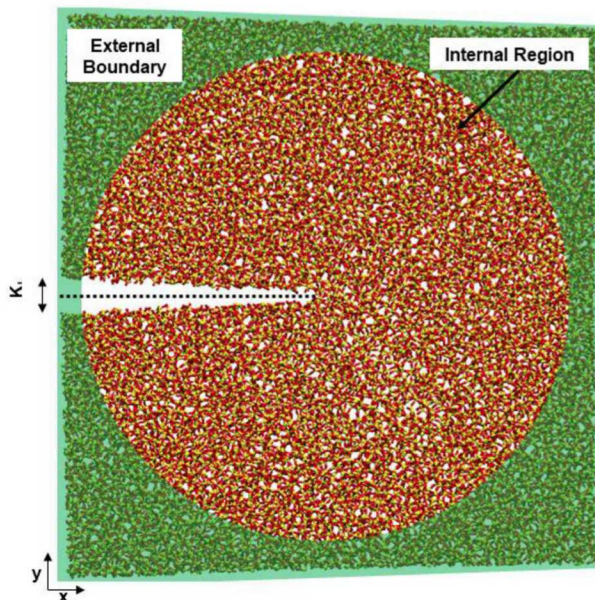


Subcritical fracture is an example of a nano-confined space – a location where the chemistry of reactions will be impacted by:

- Proximity of two surfaces
- Changes in water structure
- Changes in ion adsorption mechanisms

Fracture geometries are wedge-shaped, introducing the effects of nanoconfinement on geochemical fluids over a range of pore sizes from the tip to the bulk solution.

The chemical reactions that occur in a subcritical fracture impact the mechanical properties of the material and influence fracture propagation.



- Schematic of the quasi-2D silica system with a slit crack.
- Bonds are severed to form a slit crack.
- Atoms in the boundary region are fixed to the displacement proscribed by mode I loading (radius 3.2 nm).
- In the cylindrical region, the atoms are free to relax to a minimum energy configuration
- The axis of the cylinder is out-of-plane and the thickness of the system is 2.8 nm.

## Project Objectives



- ❖ Develop a fundamental, atomistic-level understanding of the *chemical-mechanical* processes that **control subcritical cracks** in low-permeability geomaterials.
- ❖ Link atomic-scale insight to macroscale observables.
- ❖ Address how **chemical environment** affects **mechanical behavior**.

### Definitions:

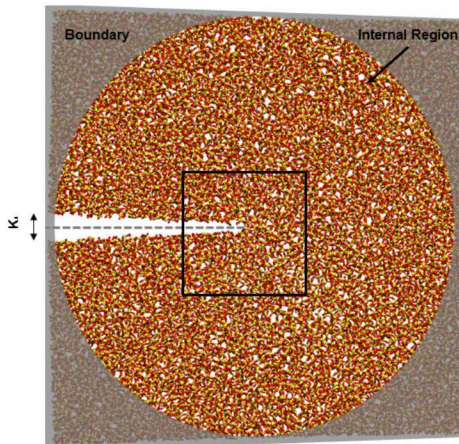
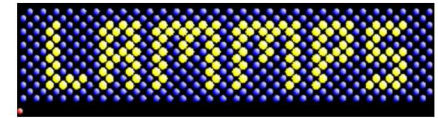
- ❑ Classical fracture mechanics: the fracture toughness,  $K_{IC}$  is the point at
- ❑ which a pre-existing fracture converts from subcritical to critical behavior.
- ❑ Our definition( $eK_{IC}$ ): the loading at which the first fracture event occurs.
- ❑ Our  $eK_{IC}$  is impacted by the chemical environment unlike the strict  $K_{IC}$  which is defined as an intrinsic material property.



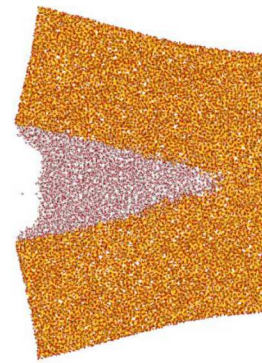
- Classical molecular dynamics for large scale simulation of silica fracture
- ReaxFF: Bond-order based forcefield including reactive water and silica bond breakage and formation (Fogarty et al. *J. Chem. Phys.* (2010), Yeon and van Duin, *J. Phys. Chem. C.* (2015))

$$E_{Total} = E_{Bond} + E_{Over} + E_{Under} + E_{LP} + E_{Val} + E_{Pen} + E_{Tors} + E_{Conj} + E_{VDW} + E_{Coul}$$

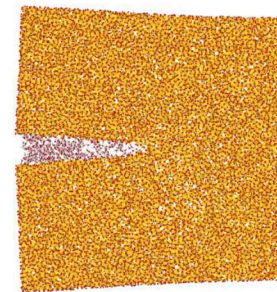
- 2D silica structures (12-replicates) were used.
- Investigated 3 different conditions to isolate chemical and mechanical effects on fracture
- Protocol: Apply initial loading (0.15 MPa/m) and relax fracture tip
  - Mechanical: increase loading (stepwise), relax for 5ps at 300K, repeat
  - Chemical-Mechanical: increase loading, add in water, relax for 5ps at 300K, repeat
  - Chemical: maintain loading, relax for 5ps at 300K, repeat



**Mechanical**  
(mechanical loading only)



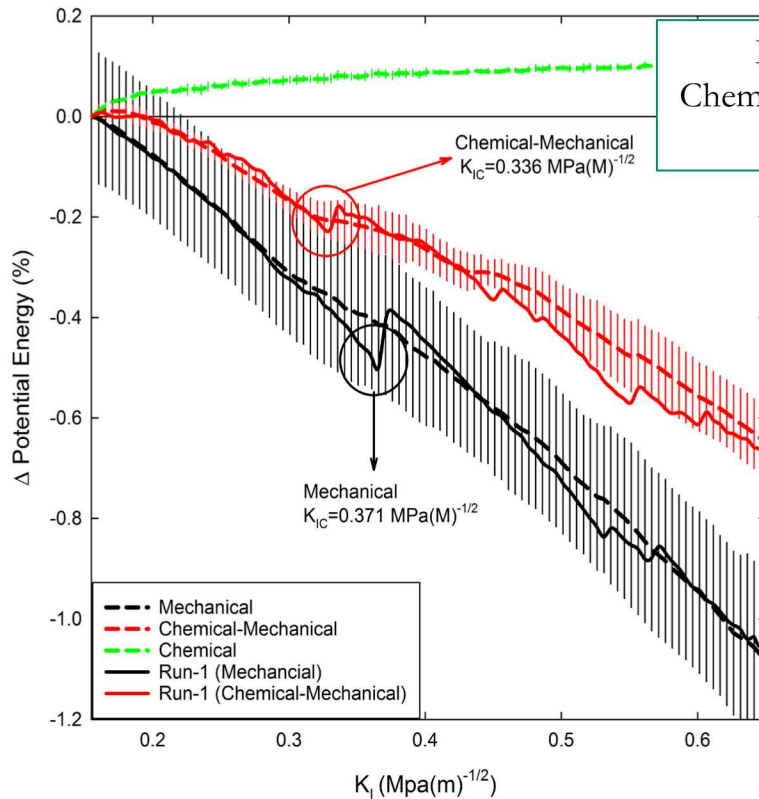
**Chemical-Mechanical**  
(aqueous environment and mechanical loading)



**Chemical**  
(aqueous environment only)

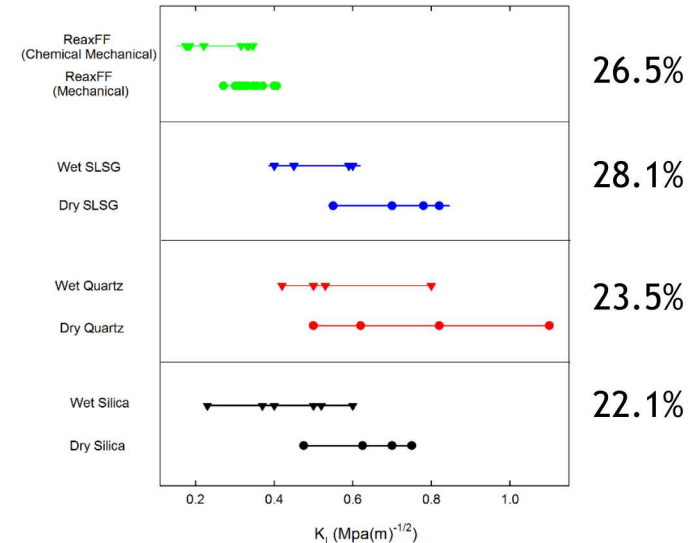
# Fracture Toughness in Vacuum and Water

- Identified from variation in the potential energy of the silica during loading
- Earlier fracture of silica in aqueous conditions
- No fracture in chemical-only systems (dissolution)
- $K_{IC}$  is lower than in experimental systems (0.78 MPa $\sqrt{m}$ ) due to resolution and temperature effects

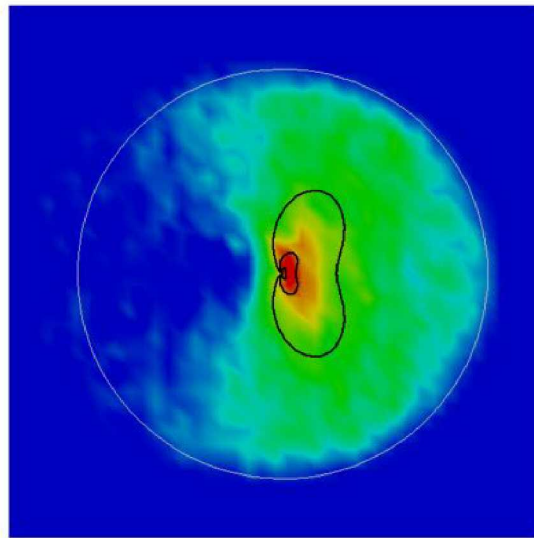


Change in potential energy for silica systems in mechanical, chemical, and chemical-mechanical conditions.

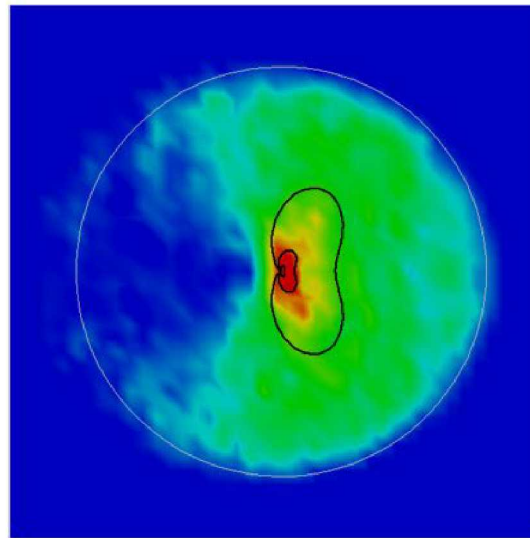
Mechanical:  $0.339 \pm 0.037$  MPa $\sqrt{m}$   
 Chemical-Mechanical:  $0.246 \pm 0.074$  MPa $\sqrt{m}$   
 Reduction in  $K_{IC}$ :  $\sim 26.5\%$



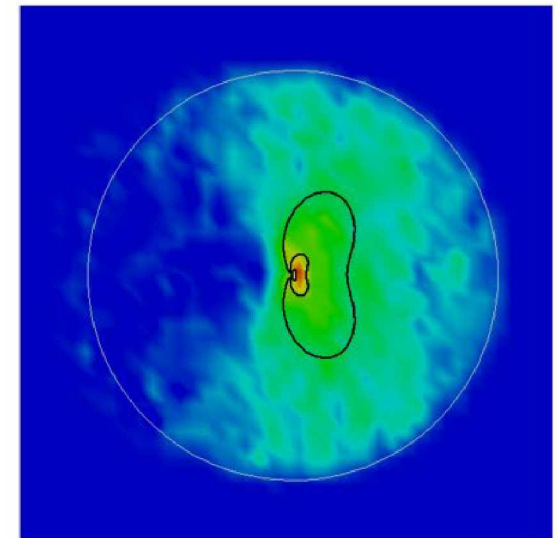
Experimental  $K_{IC}$  data for amorphous silica, quartz, and soda-lime silicate glasses in dry and aqueous environments compared with current data.



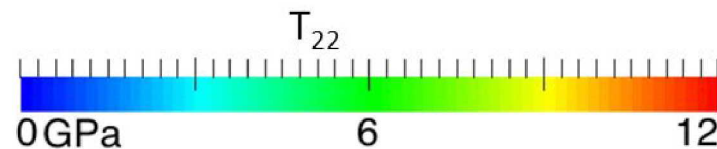
**Mechanical**  
(mechanical loading only)



**Chemical-Mechanical**  
(aqueous environment and mechanical loading)



**Chemical**  
(aqueous environment only)



Stresses from the atomistic simulations were coarse grained to describe the stress states surrounding the fracture tip. Stress fields for silica systems in mechanical ( $K_I=0.2$  MPa/ $\sqrt{m}$ ), chemical-mechanical ( $K_I=0.2$  MPa/ $\sqrt{m}$ ), and chemical conditions ( $K_I=0.15$  MPa/ $\sqrt{m}$ ).



# Energy Dissipation

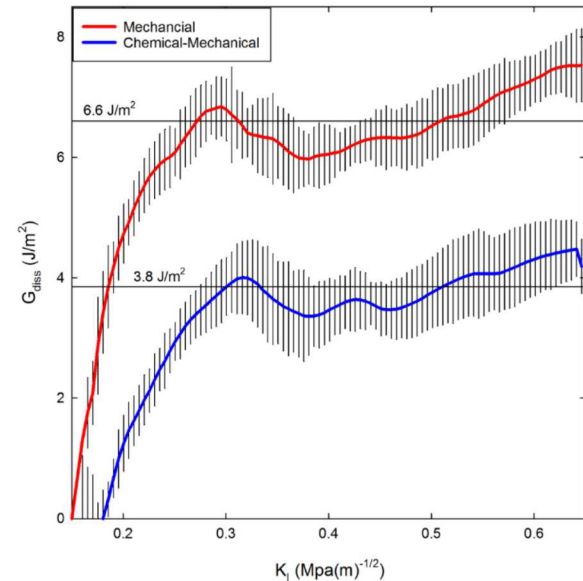
- $G$  is related to both the surface energy and dissipative energy (unrecoverable inelastic character around the fracture tip)

$$G = G_{diss} + 2\gamma_s$$

- $G_{diss}$  is calculated from energy and surface area of the fracture:

$$\frac{\Delta U}{\Delta S_A} = G_{diss}$$

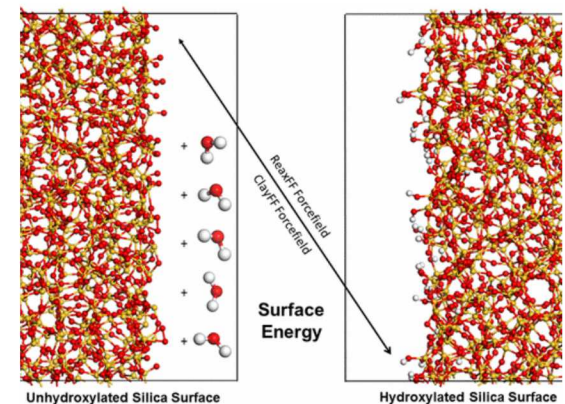
- Surface energy ( $\gamma$ ) = related to hydroxylation of the surface
- Wet fracture results in a lower  $K_{IC}$  value and lower  $G_{IC}$ , due to larger dissipation energy
- Larger  $G_{diss}$  relates to the strain distribution surrounding the fracture tip



Energy dissipation ( $G_{diss}$ ) during crack loading and subsequent crack propagation for silica systems

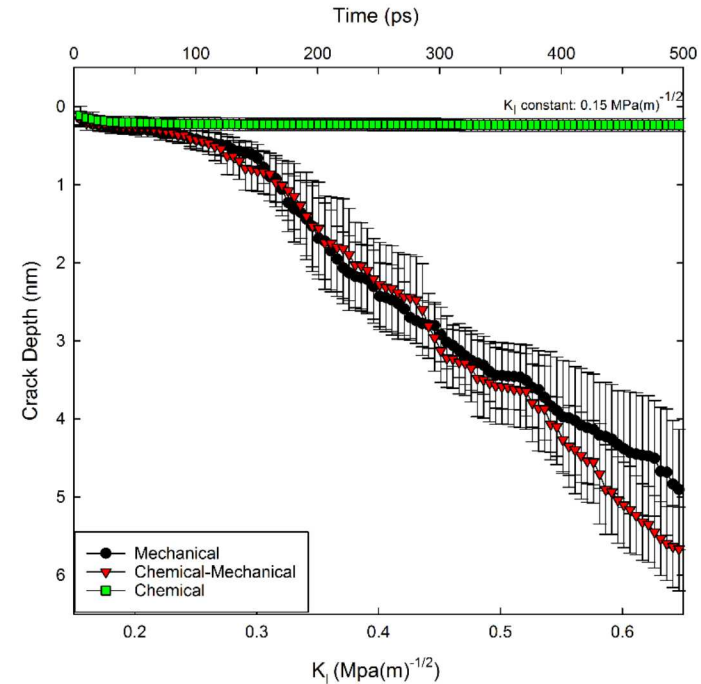
Fracture properties of silica in mechanical and chemical-mechanical conditions.

	$K_{IC}$ (MPa $\sqrt{m}$ )	$G_{IC}$ (J/m $^2$ )	$G_{diss}$ (J/m $^2$ )	Si-OH (#/nm $^2$ )	$\gamma$ (J/m $^2$ )
Mechanical	0.339 $\pm$ 0.037	8.8	6.6	0.0	1.1
Chemical-Mechanical	0.246 $\pm$ 0.074	4.6	3.8	3.1	0.4



## Fracture Propagation in Water

- ❑ Fracture depth identifies aggregate effect of aqueous environment on fracture
- ❑ Chemical-mechanical conditions: longer fracture propagation, larger number of fracture events and slightly shorter average fracture length
- ❑ Chemical effects become more prominent as the fracture propagates
- ❑ May be altering the conditions for fracture (bond stretching, stress states etc.)
- ❑ Chemical impact is more than additive on fracture growth

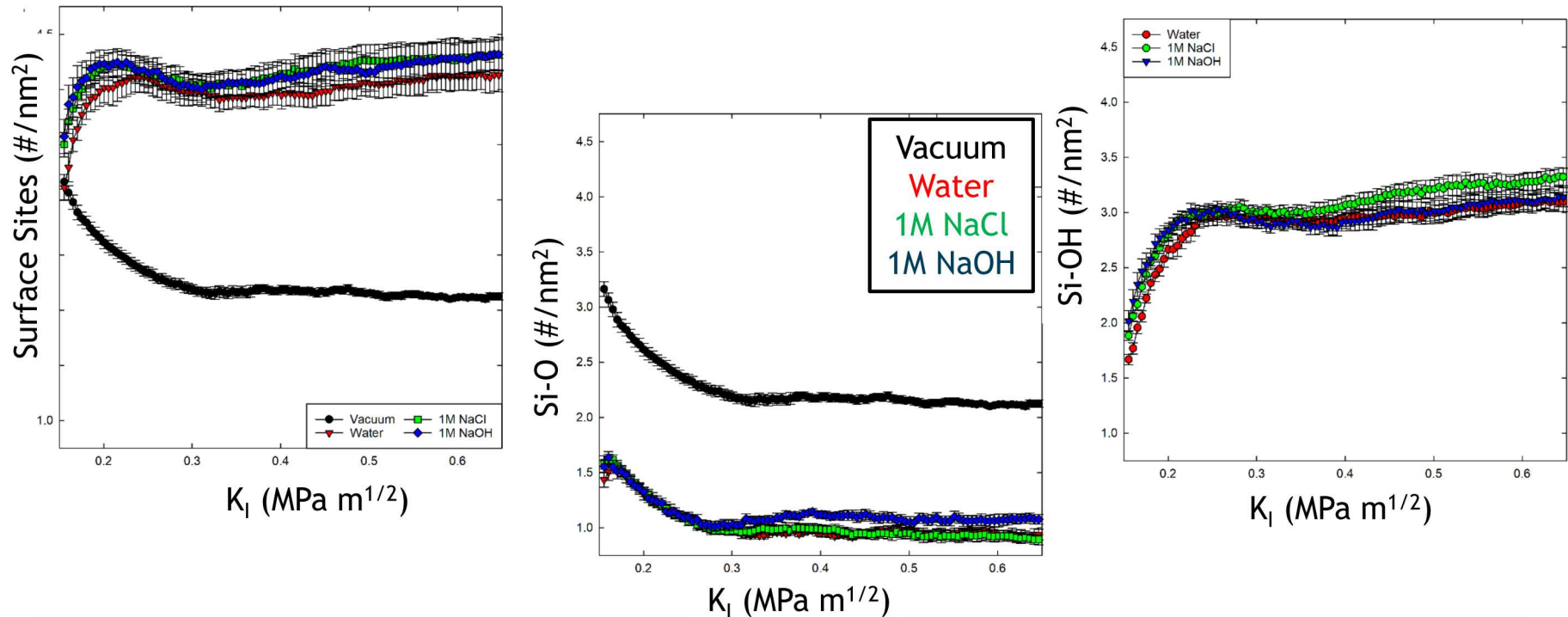


Crack depth for silica systems in mechanical, chemical, and chemical-mechanical conditions.

Crack propagation data for silica systems under different conditions.

Conditions	Propagation (nm)	Fracture Events* (#)	Average Fracture Length (nm)	Longest Fracture (nm)	Fracture Velocity (m/s)
Mechanical	$4.92 \pm 0.76$	$11.50 \pm 2.06$	$0.35 \pm 0.08$	$0.90 \pm 0.23$	$9.85 \pm 1.51$
Chemical	$0.23 \pm 0.07$	$0.50 \pm 0.50$	$0.16 \pm 0.08$	$0.10 \pm 0.08$	$0.47 \pm 0.16$
Chemical-Mechanical	$5.69 \pm 0.53$	$14.83 \pm 2.41$	$0.32 \pm 0.06$	$0.97 \pm 0.38$	$11.38 \pm 1.07$



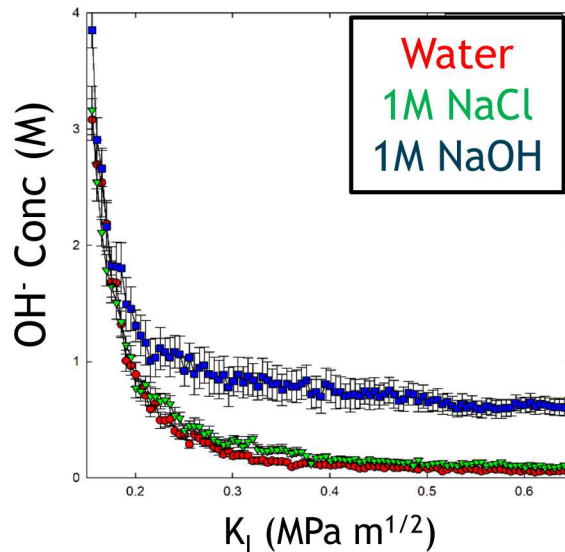


- ☐ In vacuum, there are no Si-OH surface sites.
- ☐ In vacuum, surface relaxation causes some surface sites to reconnect and form siloxane bonds.
- ☐ Silica surfaces exposed to salt solutions exhibit more surface sites than those in pure water suggesting a higher concentration of broken Si-O- bonds.
- ☐ More Si-OH sites form in 1M NaCl solutions; more Si-O<sup>-</sup> sites form in 1M NaOH solutions.
- ☐ Surface structure is influenced by solution composition.

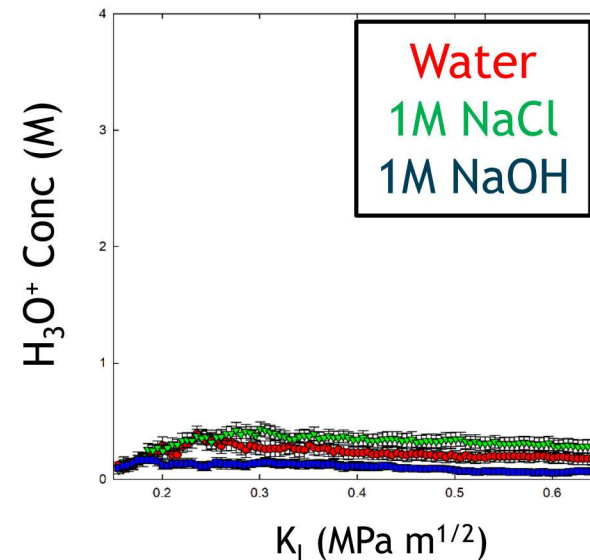
# Solution Composition in Fracture



OH<sup>-</sup> Concentration in Solution



H<sub>3</sub>O<sup>+</sup> Concentration in Solution

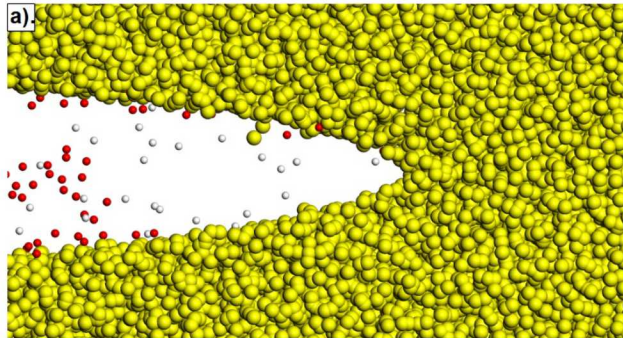


- ❑ Rapid change in concentrations occurs with initial loading before crack propagation
- ❑ Steady-state concentration occurs at  $\sim 0.25\text{-}0.3 \text{ MPa}\sqrt{m}$  due to balance of rate of water infiltration and addition of NaCl or NaOH molecules as fracture is loaded
- ❑ Concentration of H<sub>3</sub>O<sup>+</sup> increases with decreasing pH: 1 M NaOH < water < 1 M NaCl.
- ❑ Silica dissolution should be higher in both NaCl and NaOH solutions than in pure water.

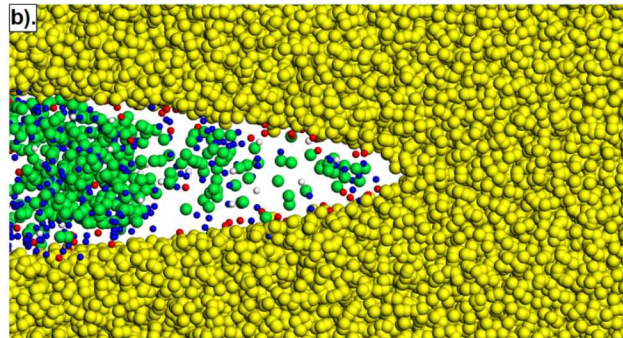
# Accessibility of Fracture Tip to Different Ions



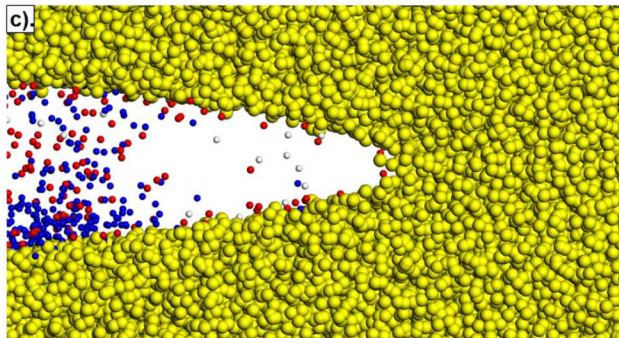
Water



1M NaCl



1M NaOH

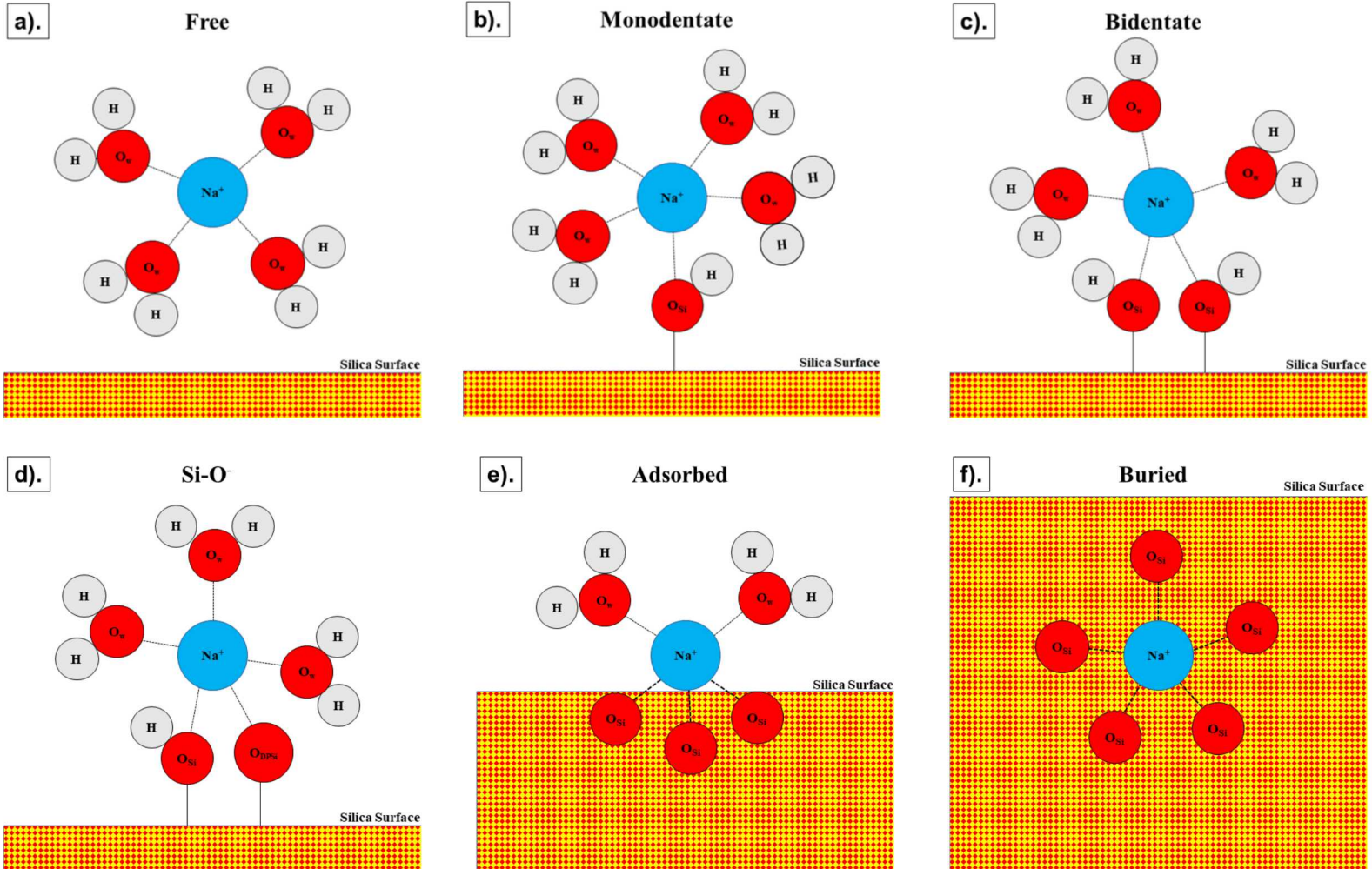


- ❑ NaCl and NaOH molecules were added at the widest point in the fracture to emulate diffusion of ions from the bulk fluid
- ❑ For silica fracture in  $\text{H}_2\text{O}$ , the crack tip is filled with  $\text{H}_2\text{O}$  and a few  $\text{H}_3\text{O}^+$
- ❑  $\text{OH}^-$  migrates to fracture tip in both NaCl and NaOH solutions
- ❑ In NaCl solutions, the tip is filled with  $\text{Na}^+$ ,  $\text{Cl}^-$  and some  $\text{OH}^-$ ,  $\text{H}_3\text{O}^+$
- ❑ Limited  $\text{Na}^+$  diffusion into crack tip from NaOH solution; crack tip contains surface coordinated  $\text{OH}^-$  or free  $\text{H}_3\text{O}^+$ .





# Na<sup>+</sup> Coordination Structures



Structure (d) does not occur on flat surfaces: nanoconfinement effect of fracture tip

# Fracture Properties of Silica for Different Environmental Conditions

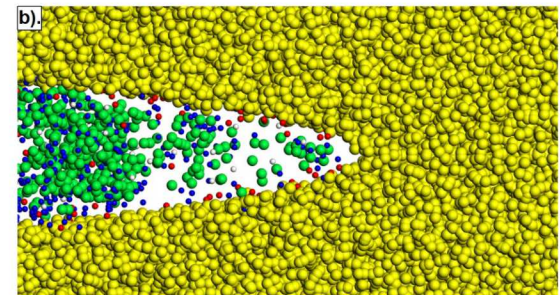
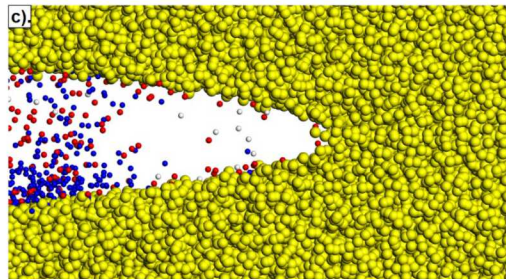
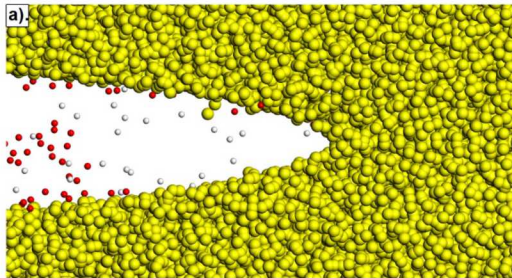


	eK <sub>IC</sub> (MPa√m)		Fracture Events (#)	G <sub>IC</sub> (J/m <sup>2</sup> )	G <sub>diss</sub> (J/m <sup>2</sup> )	Si-OH (#/nm <sup>2</sup> )	γ (J/m <sup>2</sup> )
	First	Average					
Vacuum	0.34±0.04	0.43±0.04	3.67±1.18	7.91	6.78	0.00	1.13
Water	0.20±0.06	0.37±0.05	4.33±1.03	4.59	4.21	3.10	0.38
1M NaCl	0.28±0.09	0.41±0.05	5.42±1.66	5.14	4.75	3.04	0.39
1M NaOH	0.19±0.05	0.37±0.05	6.00±1.41	5.47	5.06	2.95	0.41

## Ranking of factors that influence environmentally assisted fracture

	eK* <sub>IC</sub>	Fracture events	Dissolution	Si-O <sup>-</sup> #	Tip access	Radius of curvature
Water	2	3	3	3	3	3
1M NaCl	1	2	1	2	1	1
1M NaOH	3	1	2	1	2	2

- ❖ Amorphous silica is substantially weaker when in contact with aqueous solutions than in vacuum due to chemical reactions with preexisting cracks.
- ❖ Fracture toughness is lowest for silica in 1M NaOH solutions. The basic solution leads to higher surface deprotonation, less dissolution, and a narrower radius of curvature than in an acidic environment.
- ❖ The 1M NaCl solution causes more silica dissolution than pure water or a 1M NaOH solution and changes the geometry of the fracture tip. In addition  $Q^0$  silica species are observed in solution.
- ❖ The nanoconfinement at the fracture tip results in different  $Na^+$  adsorption mechanisms than on a flat surface.



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