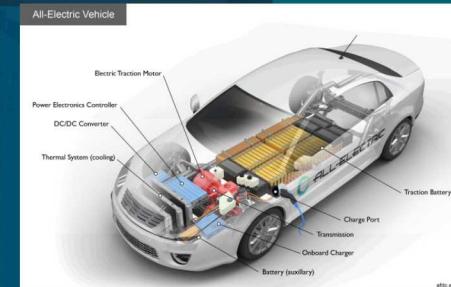
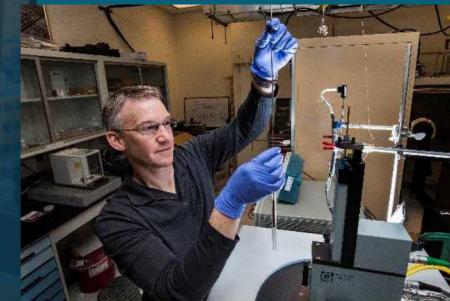


# Next Generation Soft Magnetic Materials



Todd C. Monson, Tyler E. Stevens, Charles J. Pearce, Mark A. Rodriguez, Stanley Atcity

# Soft magnetic materials are ubiquitous

**Satellites**



**Electric ships**



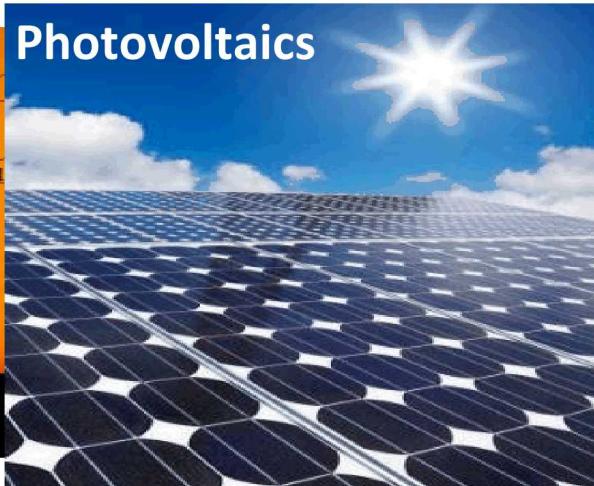
**UAVs**



**Transmission**



**Photovoltaics**



**Electric vehicles**



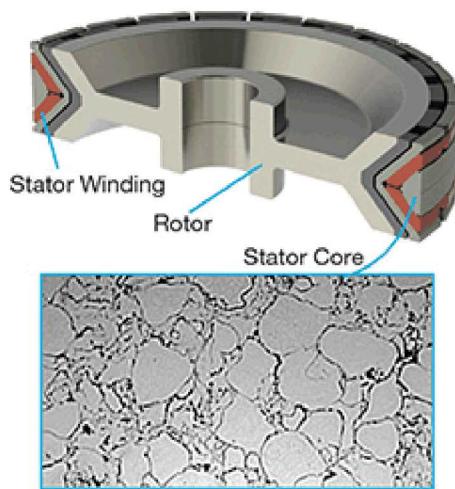
Courtesy of Bob Kaplar

# Soft Magnetic Materials and Energy

## Inductor cores



## Transformer cores

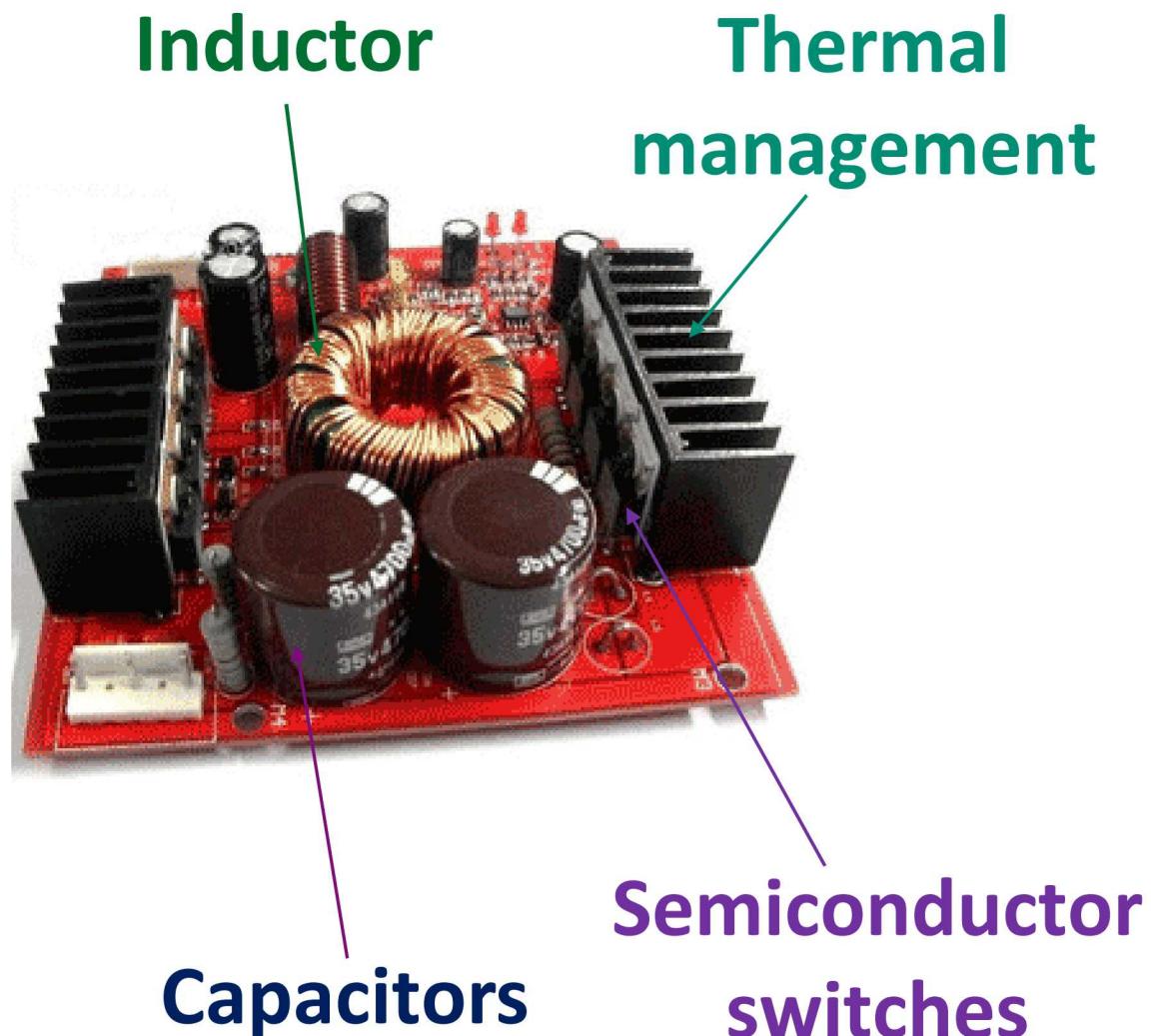


## Electrical machines (motors & generators)

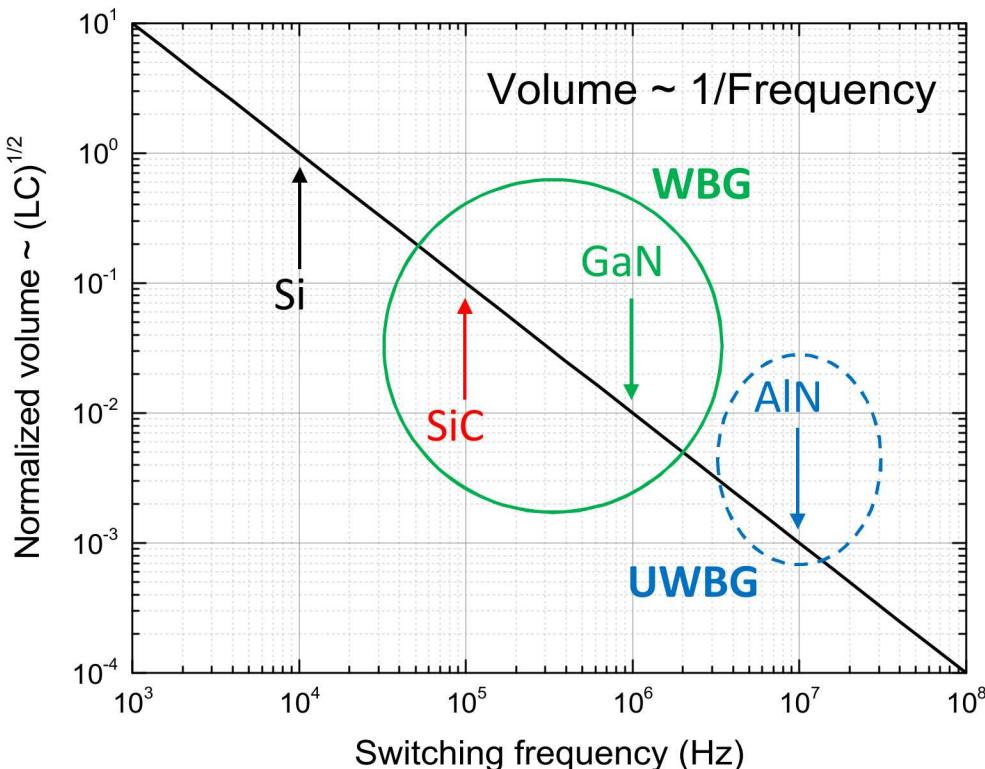
# Magnetics Impact Power System Volume and Weight

***WBG/UWBG  
materials enable  
higher switching  
frequency and better  
thermal management***

**Passive elements and  
thermal management  
comprise the bulk of  
the volume and mass  
of a power converter**



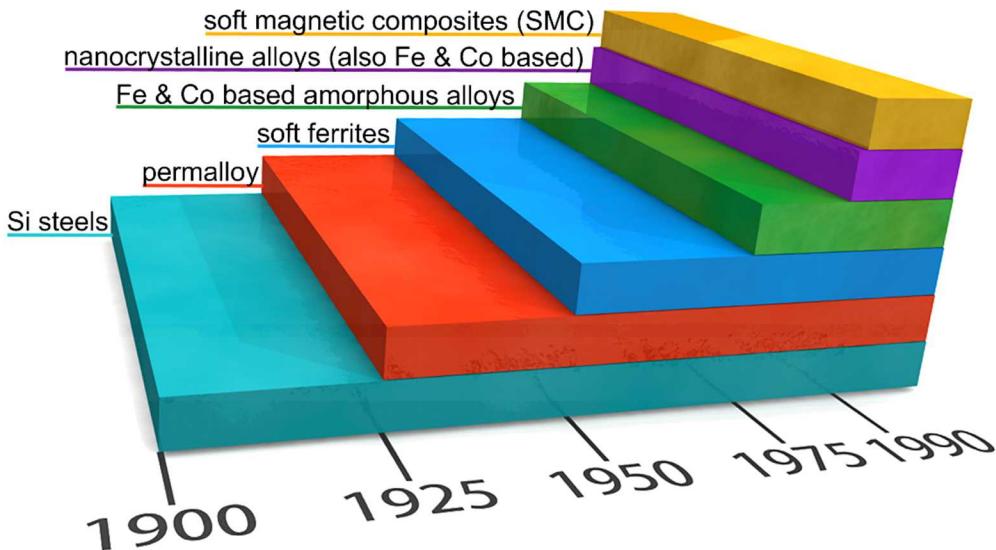
# Higher Frequencies Decrease Inductance Requirements, However...



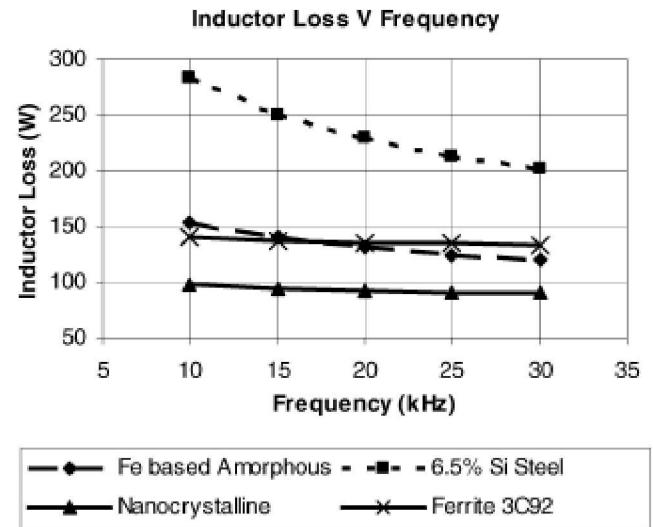
- Higher switching frequency is enabled by scaling properties of WBG/UWBG materials
- Ideal  $1/f$  SWaP dependence (true dependence likely weaker  $1/f^n$  with  $n < 1$  due to nonidealities)
- Other benefits of WBG/UWBG exist, e.g. higher voltage without series stacking of devices, and higher temperature operation

***Inductive core materials have essentially been an afterthought and new magnetic materials are needed***

# Soft magnetic material development



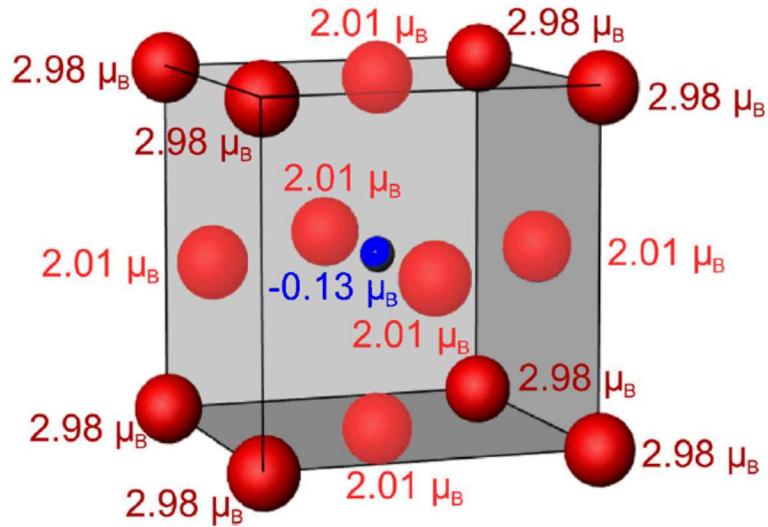
Adapted from: L.A. Dobrzański, M. Drak, B. Ziębowicz, Materials with specific magnetic properties, Journal of Achievements in Materials and Manufacturing Eng., 17, 37 (2006).



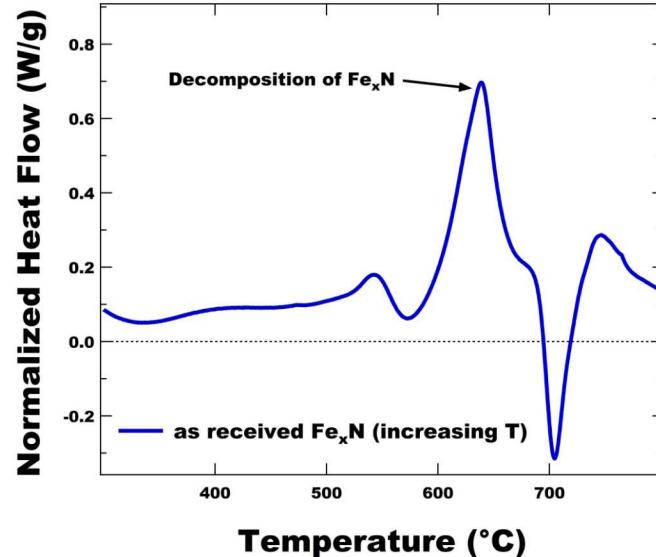
B.J. Lyons, J.G. Hayes, M.G. Egan, Magnetic Material Comparisons for High-Current Inductors in Low-Medium Frequency DC-DC Converters, IEEE, 71 (2007).

Magnetic Material	$J_s$ (T)	$\rho(\mu\Omega\cdot m)$	Cost
VITROPERM (Vacuumschmelze)	1.20	1.15	High
Metglas 2605SC	1.60	1.37	High
Ferrite (Ferroxcube)	0.52	$5 \times 10^6$	Low
Si steel	1.87	0.05	Low
CoFeV (Supermendur)	2.40	0.08	Med
$\gamma'$ -Fe <sub>4</sub> N	1.89	> 200	Low
CoFeP	~2.30	> 100	Med

# $\gamma'$ -Fe<sub>4</sub>N



**fcc  $\gamma$ Fe structure stabilized by interstitial nitrogen in the body center**

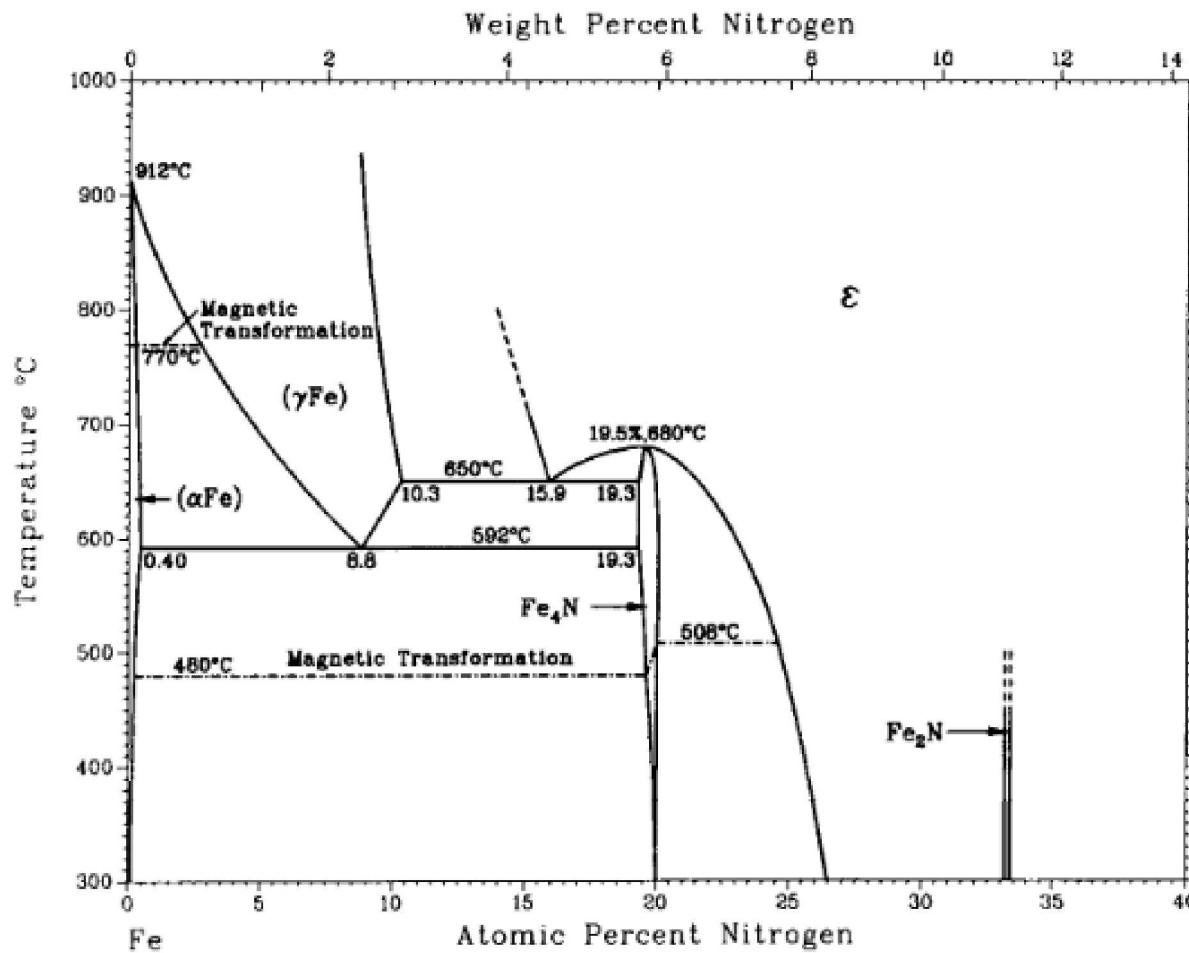


**Relatively low thermal decomposition limits consolidation & fabrication methods**

G. Scheunert, et al., A review of high magnetic moment thin films for microscale and nanotechnology Applications, *Appl. Phys. Rev.*, 3, 011301 (2016).

J.M.D. Coey, *Magnetism and Magnetic Materials* (Cambridge University Press, Cambridge, UK, 2012).

# Iron nitride phase diagram



H.A. Wriedt, N.A. Gokcen, and R.H. Nafziger, 1987.

Commercial synthesis yields mixture of  $\text{Fe}_3\text{N}$  and  $\text{Fe}_4\text{N}$

# Previous Syntheses of $\gamma'$ - $\text{Fe}_4\text{N}$

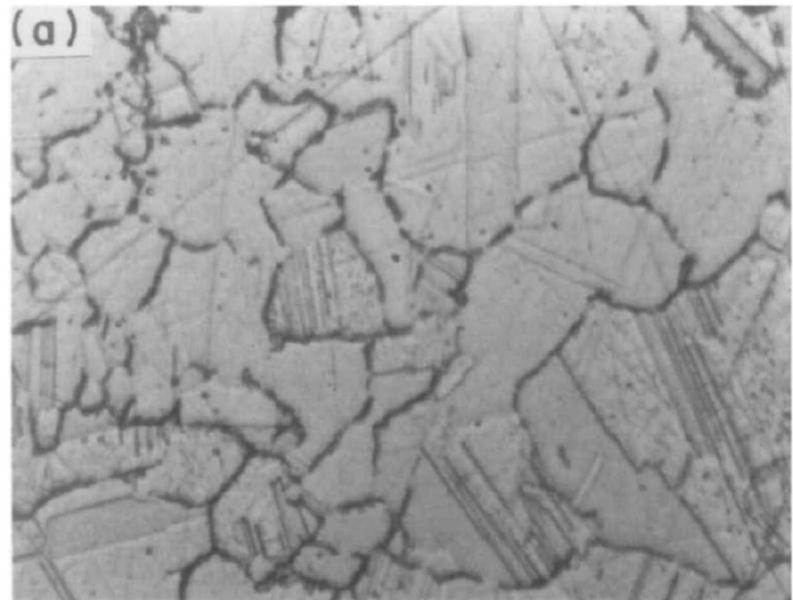
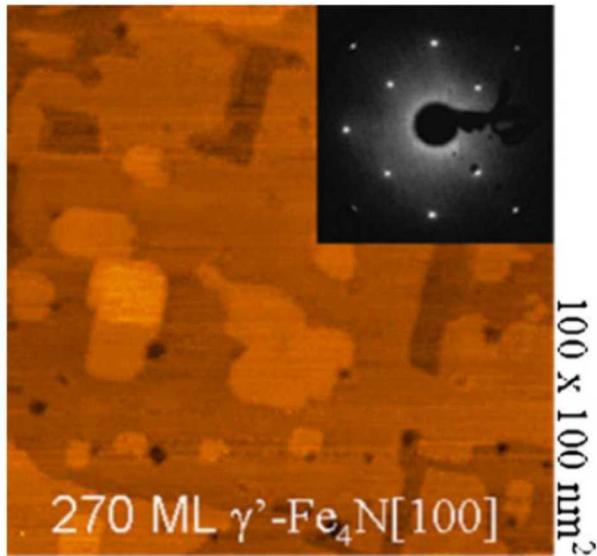


Fig. 1. STM image of a 270 monolayers (ML) thick  $\gamma'$ - $\text{Fe}_4\text{N}$  film grown on Cu(100). The inset shows the corresponding LEED pattern (110 eV).

D. Ecija, et. al., "Magnetisation reversal of epitaxial films of  $\gamma'$ - $\text{Fe}_4\text{N}$  on Cu(100)", J. Magn. Mag. Mat., 316, 321 (2007).

S.K. Chen, et. al., "Synthesis and magnetic properties of  $\text{Fe}_4\text{N}$  and (Fe, Ni)<sub>4</sub>N sheets", J. Magn. Mag. Mat., 110, 65 (1991).

**Up to 50 nm thick**

**25  $\mu\text{m}$  foils**

# $\gamma'$ -Fe<sub>4</sub>N synthesis and processing

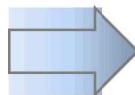


**U.S. Patent # 9,963,344**

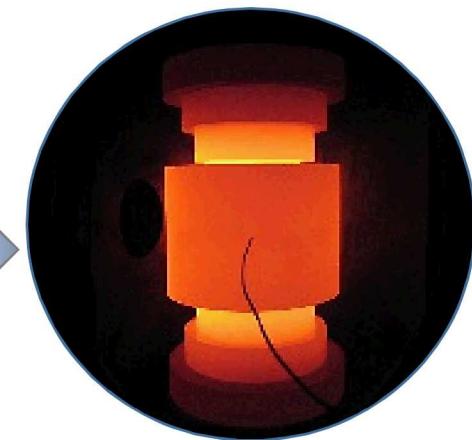
Atomization  
of powder



Fluidized Bed Furnace



SPS Consolidation



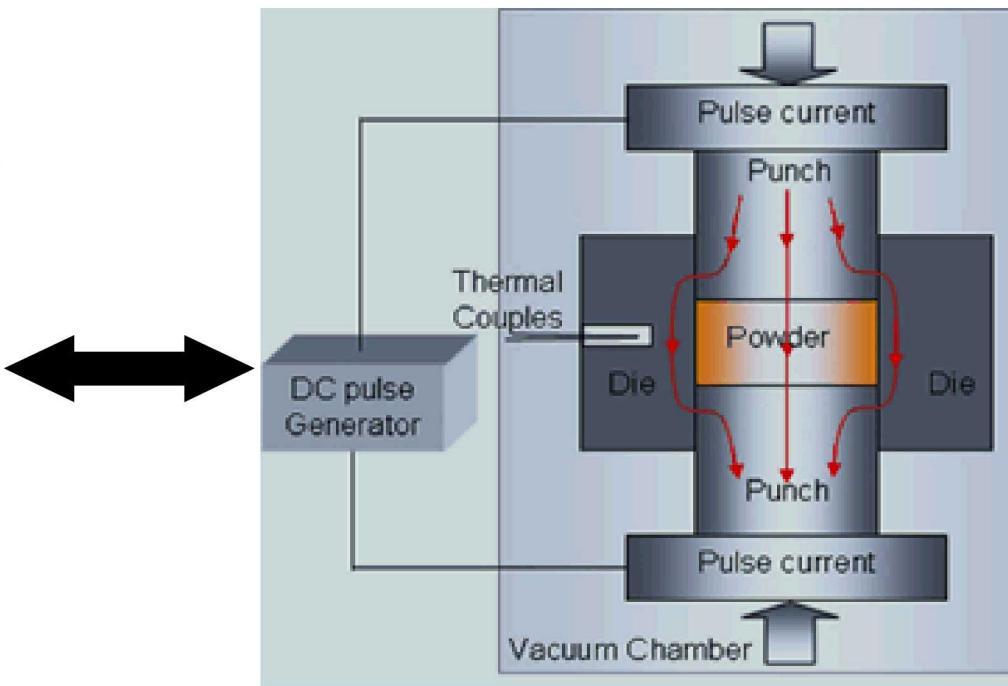
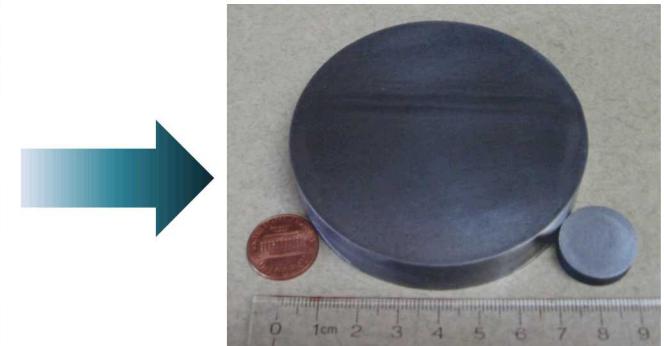
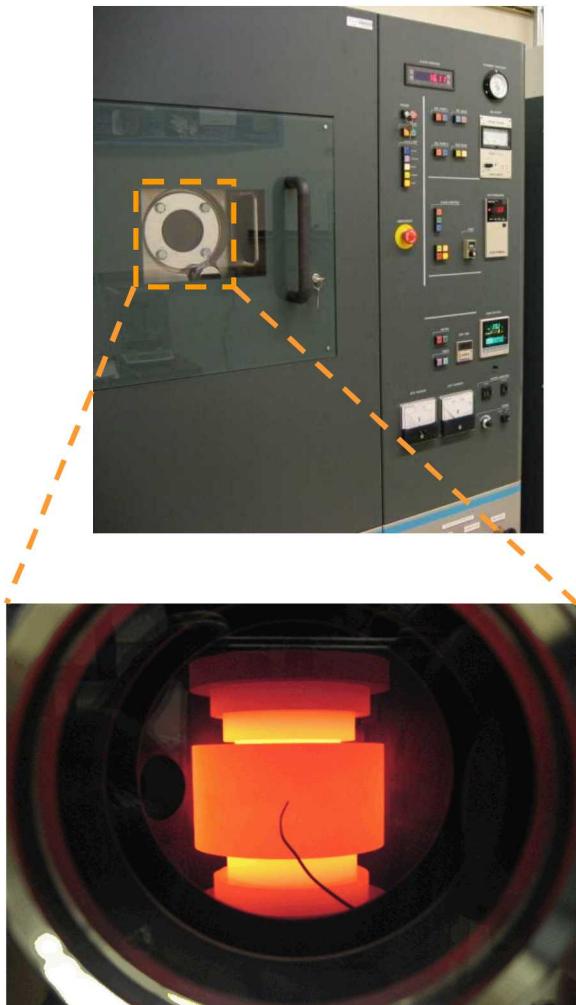
Spark Plasma Sintering  
Fast sintering

Cryomilling  
Severe Plastic  
deformation



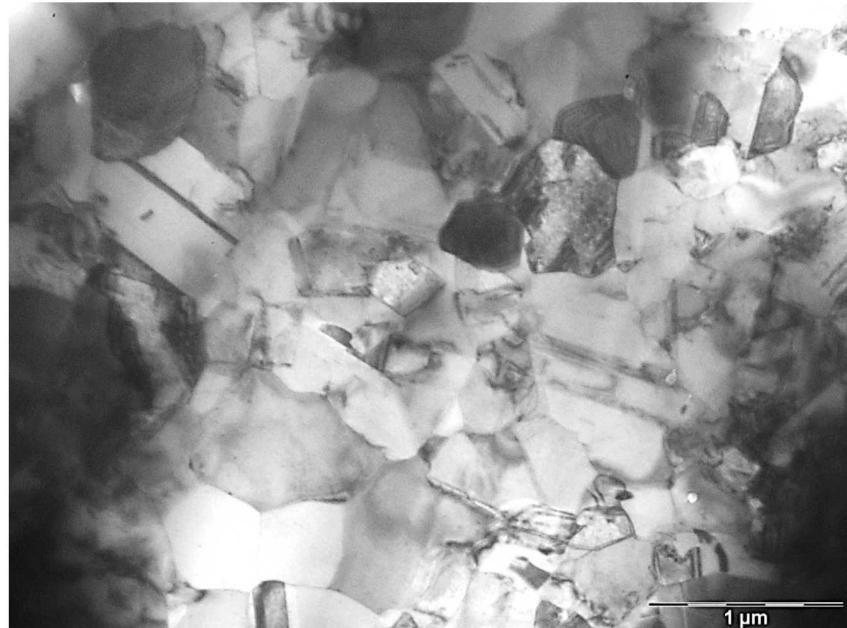
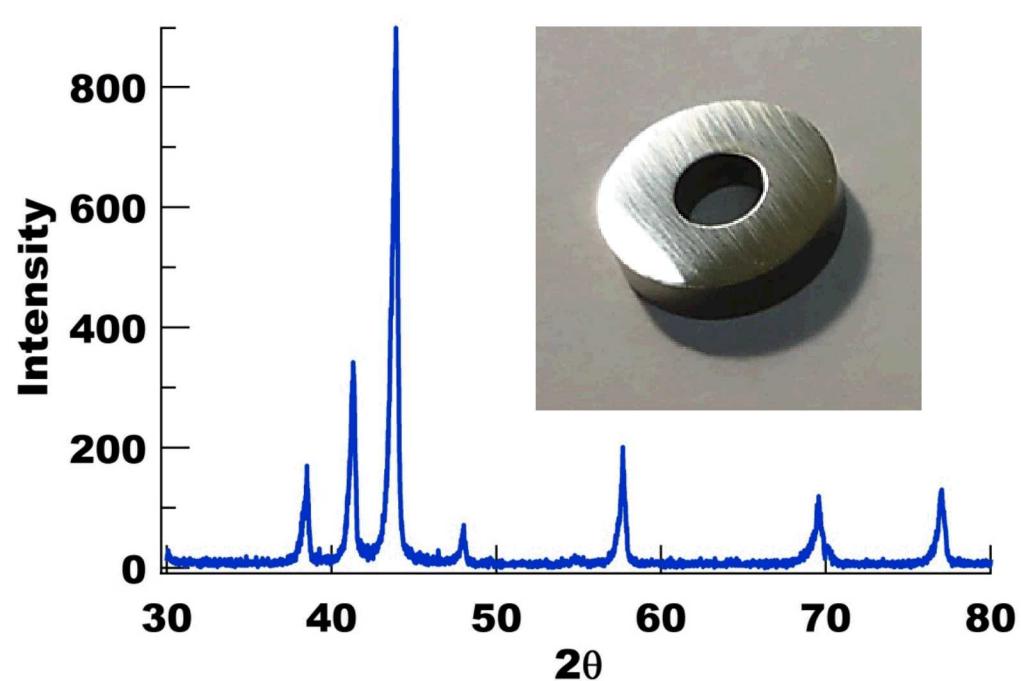
- Pressure and pulsed current assisted sintering process
- Precision control over heat, pressure, and time
- Restrain grain growth
- Full densification

# Spark plasma sintering (SPS)



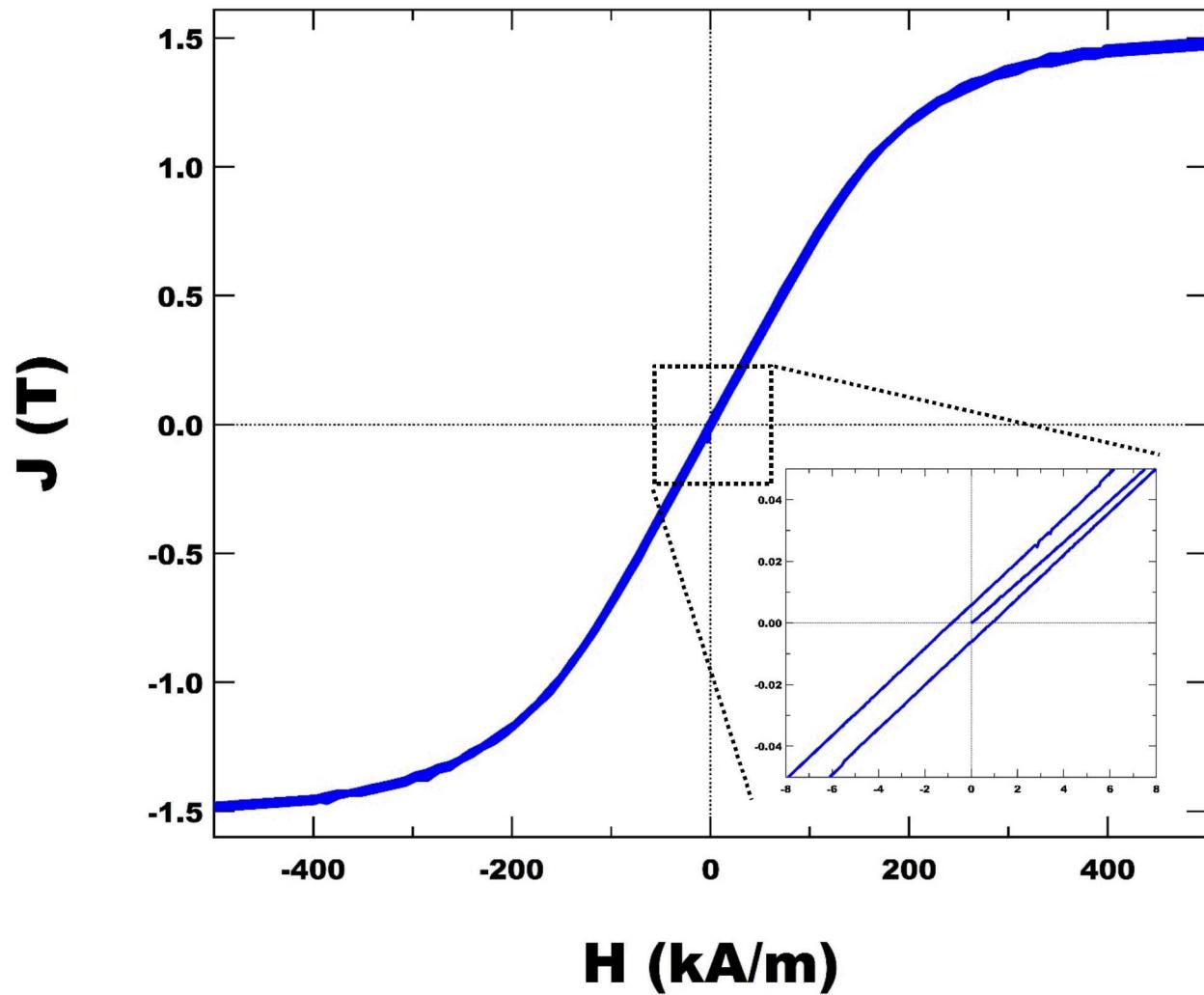
# SPS consolidated iron nitride

First ever bulk  $\gamma'$ - $\text{Fe}_4\text{N}$ !



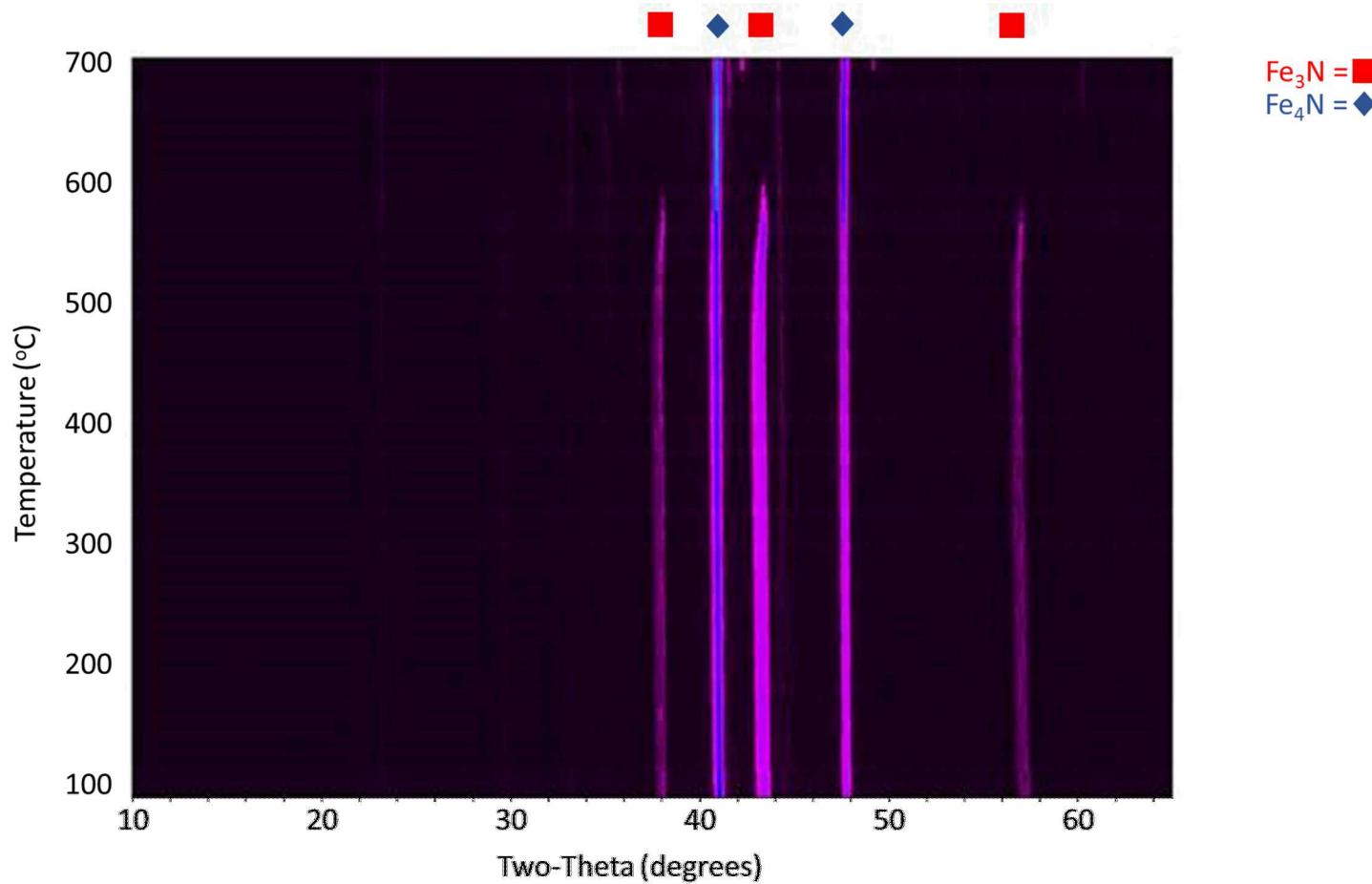
- Fe nitride powders well consolidated with little porosity
- Grain sizes 200 nm – 1  $\mu\text{m}$   $\rightarrow$  fine grain size = low loss
- $\gamma'$ - $\text{Fe}_4\text{N}$  primary phase
- $\text{Fe}_3\text{N}$  secondary phase from mixed phase starting material

# Magnetic characterization



- SPSed at  $550^{\circ}\text{C}$  and  $100 \text{ MPa}$
- $J_s = 1.62 \text{ T}$
- Theoretical  $J_s = 1.89 \text{ T}$  (SiFe is  $1.87 \text{ T}$ )
- $H_c < 1000 \text{ A/m}$

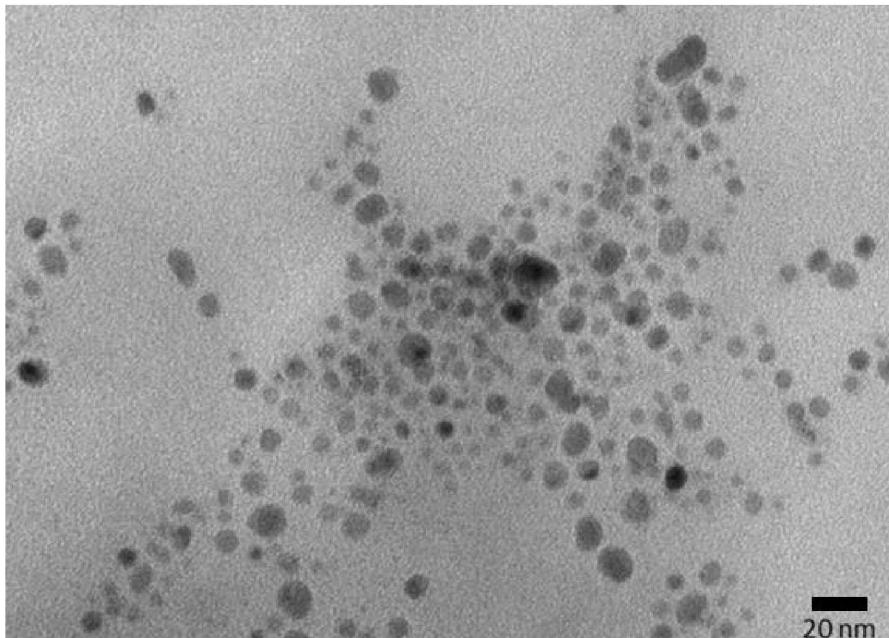
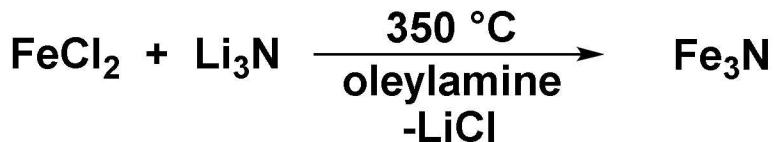
# Production of phase pure $\gamma'$ - $\text{Fe}_4\text{N}$ powder



- Simple heat treatment converts mixed phase commercial powder
- Only phase pure  $\gamma'$ - $\text{Fe}_4\text{N}$  remains

# An example of other routes to iron nitrides

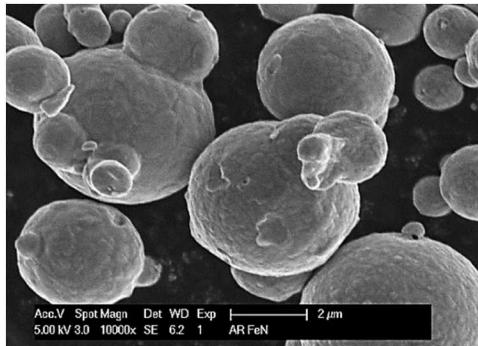
Reaction of  $\text{FeCl}_2$  with  $\text{Li}_3\text{N}$  results in  $\text{Fe}_3\text{N}$



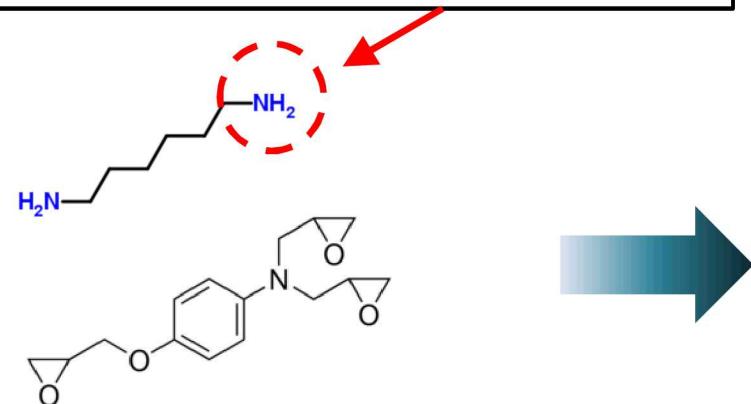
- 0.124 M  $\text{FeCl}_2$  oleylamine
- Injected at 80  $\mu\text{L}/\text{min}$
- 4 hour reaction time
- Improvements to size distribution possible

# Iron Nitride/Epoxy Composites

Convert commercial  $\text{Fe}_x\text{N}$  powder to phase pure  $\text{Fe}_4\text{N}$

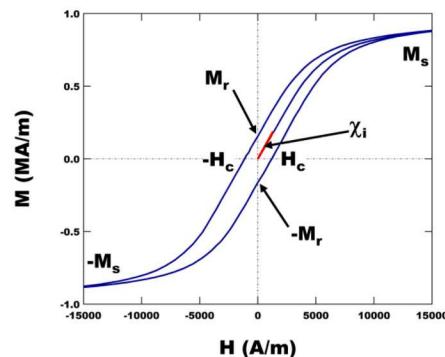
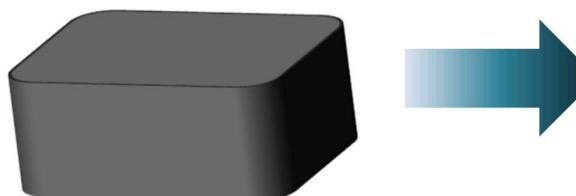


Diamines will bond directly to  $\text{Fe}_4\text{N}$  surface and epoxy matrix for enhanced mechanical robustness and particle electrical isolation



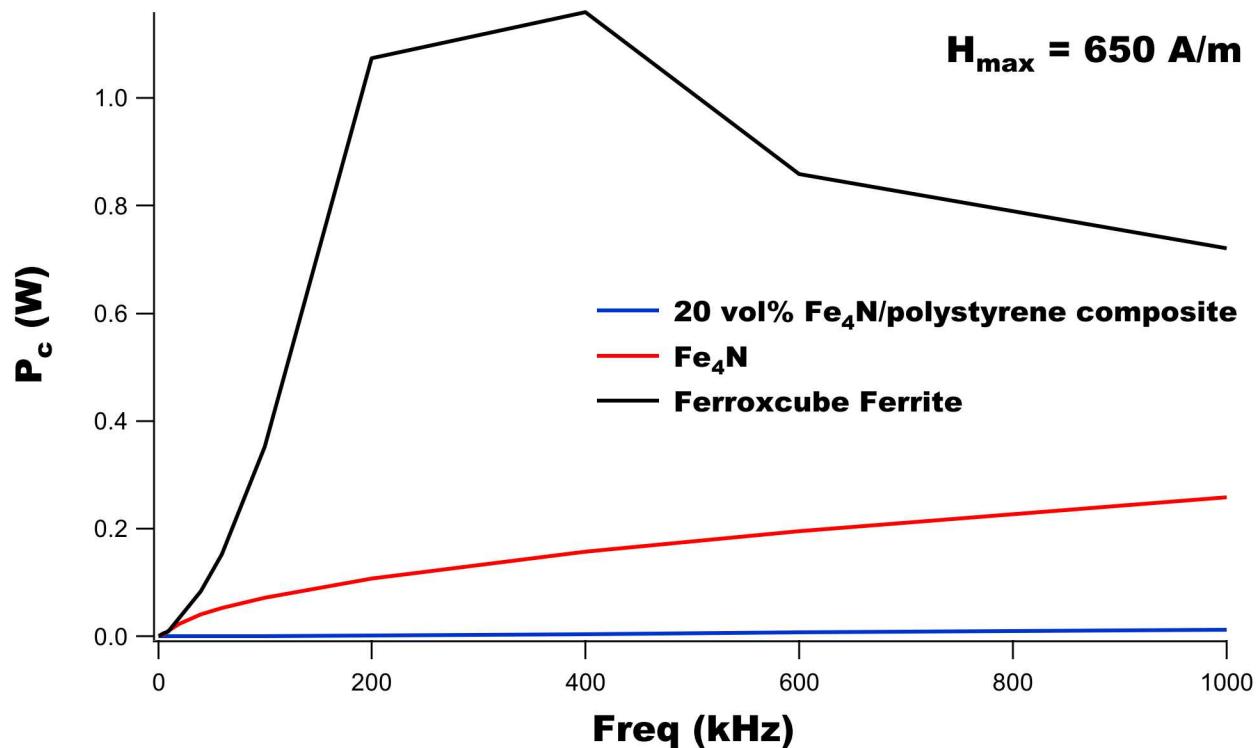
Coat  $\text{Fe}_4\text{N}$  and mix with epoxy monomers

- Pour into 3D printed mold and cure into stator/rotor part
- Press if necessary to increase density and loading factor
- Results in a net-shaped part (no machining required)



Evaluate and test

# Preliminary $\gamma'$ -Fe<sub>4</sub>N composite results



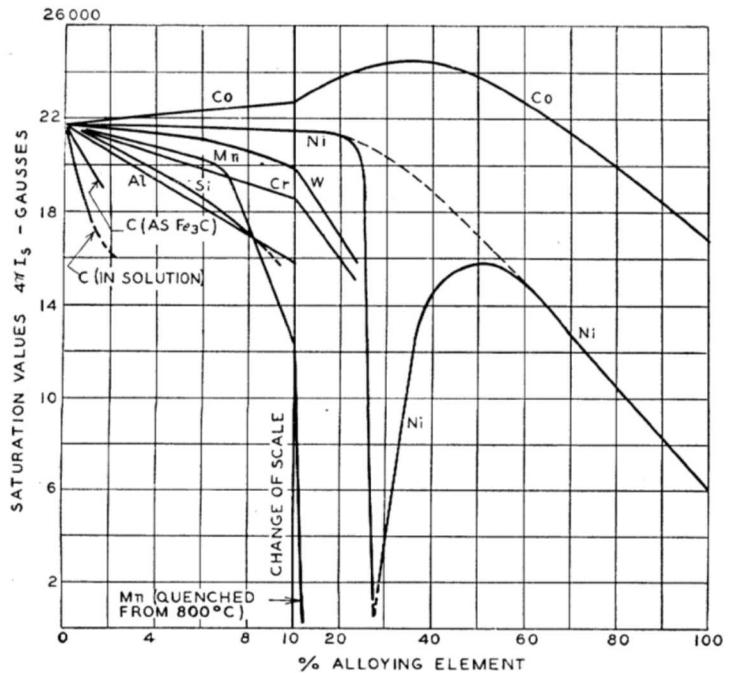
- Significantly lower core losses in Fe<sub>4</sub>N composites when compared to both bulk Fe<sub>4</sub>N and COTS ferrites
- Much higher volume loadings of Fe<sub>4</sub>N still possible
- Transitioning to epoxy matrix will exceed temperature and mechanical strength requirements

# CoFeP synthesis

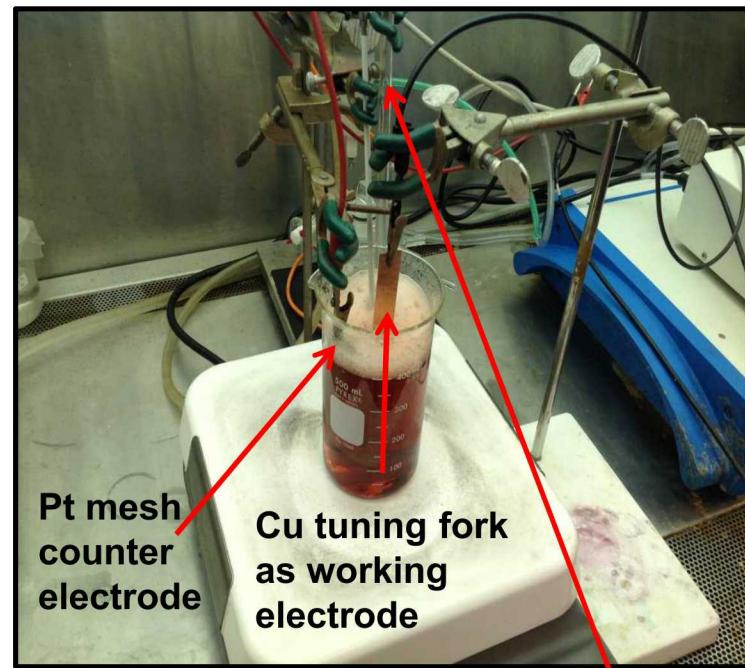
Jamin Pillars (SNL), Eric Langlois (SNL)

Figure 1. Effect of elements on the saturation values of iron

STANLEY IS ALSO WITH THE  
Corporation, East Pittsburg  
\*  $I_s$  = saturation intensity c

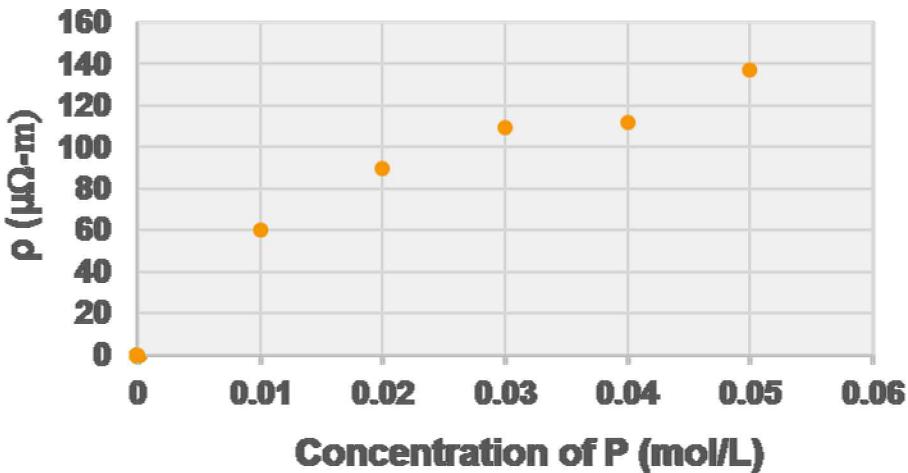


J.K. Stanley, T.D. Yensen, "Hiperco—A Magnetic Alloy," AIEE Transactions, 66, (1947).



Bubbler with  $N_2$  gas

## CoFe Phosphorous Incorporation



# Acknowledgements

- **$\gamma$ '-Fe<sub>4</sub>N R&D supported by Dr. Imre Gyuk and the Energy Storage Program in the Office of Electricity Delivery and Energy Reliability at the US Department of Energy**
- **Additional support: DOE/VTO, SNL LRD**
- **We thank Robert Delaney (Univ. of NM) for his assistance with magnetic data analysis**



# Iron nitride offers superior magnetic properties

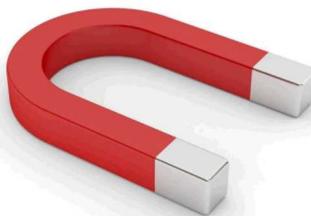
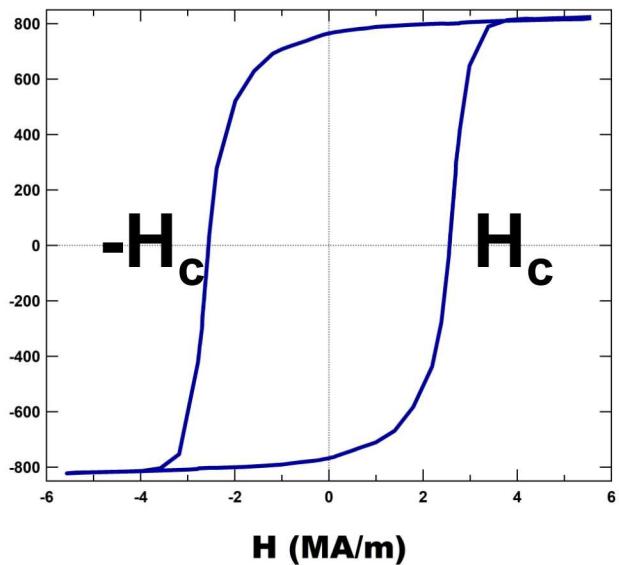
Material	$\sigma_s$ (Am <sup>2</sup> /kg)
$\alpha$ -Fe	218
Magnetite	80 - 103
FeN	209
Fe <sub>2</sub> N	~70
Fe <sub>3</sub> N	144
Fe <sub>4</sub> N	209

- Iron nitrides offer superior magnetic properties compared to oxides
- Numerous applications could benefit a straight forward route to synthesis these materials, particularly in phase pure form
- Iron nitrides are metastable, therefore synthesis is challenging

# A brief magnetics overview- hard vs. soft magnets

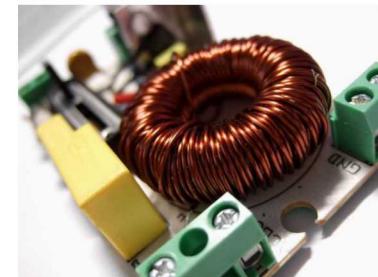
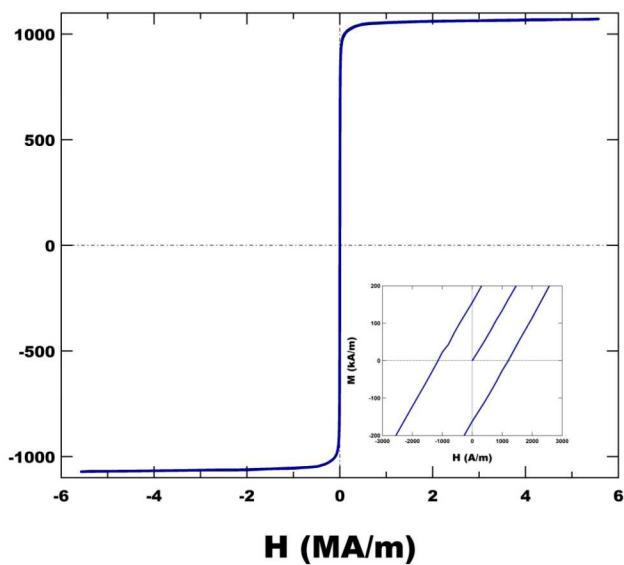
## Hard (permanent)magnet

$M$  (kA/m)

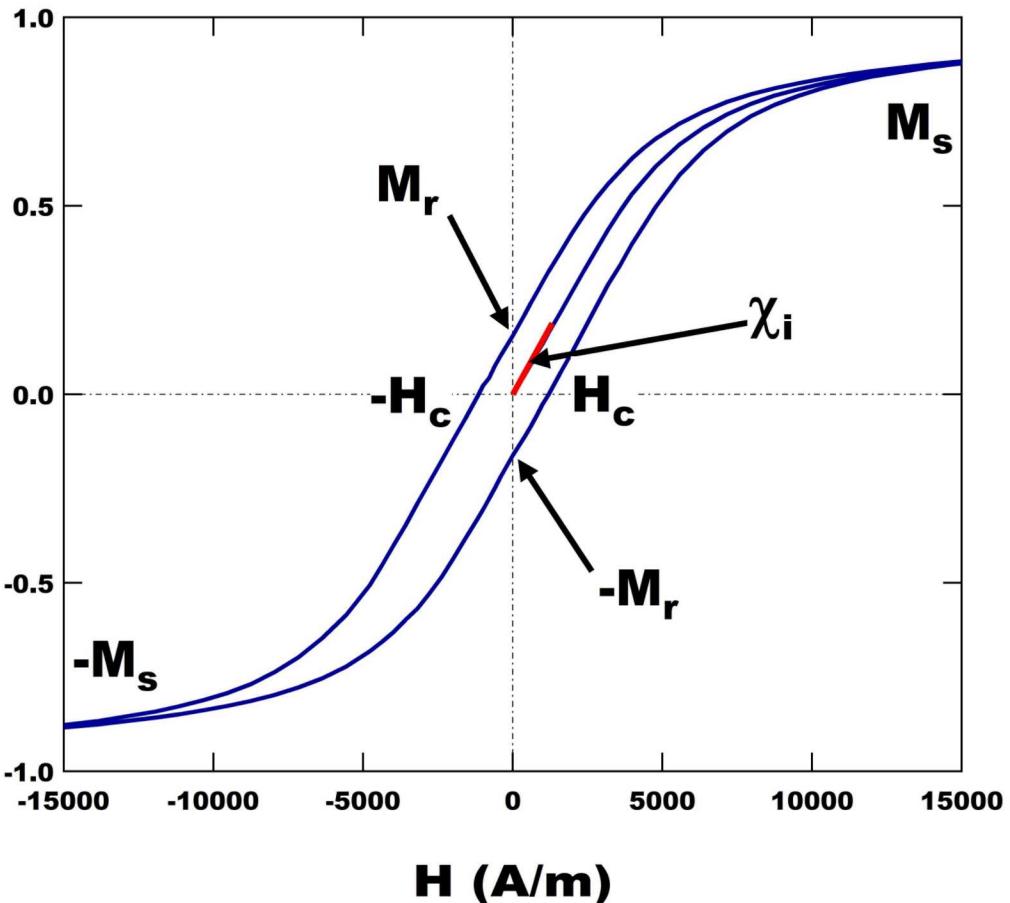


## Soft magnet

$M$  (kA/m)



# Magnetic properties



**$M_s$  = saturation magnetization**

**$M_r$  = magnetic remanence**

**$H_c$  = coercivity**

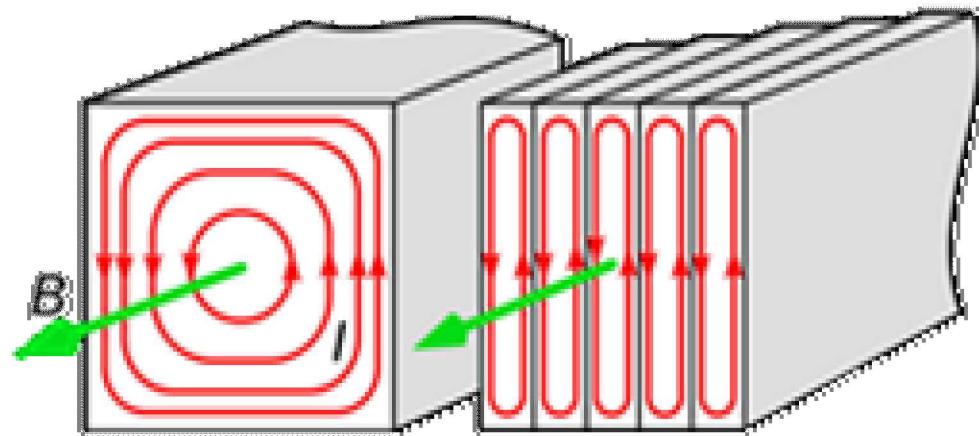
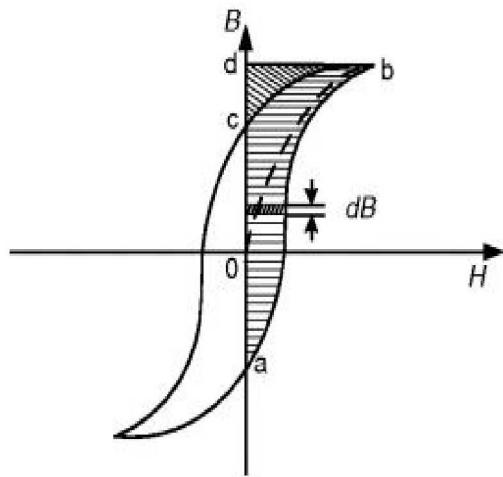
**$\chi_i$  = initial susceptibility**

**$\mu$  = permeability**

**$\mu_r$  = permeability**

$$\mu_r = \mu/\mu_0 = 1 + \chi$$

# Magnetic losses



$$\frac{P_{\text{phys}}}{V} = \oint H(t) dB$$

$$\frac{P_{\text{eddy}}}{V} = \frac{\omega B^2 A}{48\rho}$$

**Hysteresis**

**Eddy Currents**

# Coercivity as a function of particle size

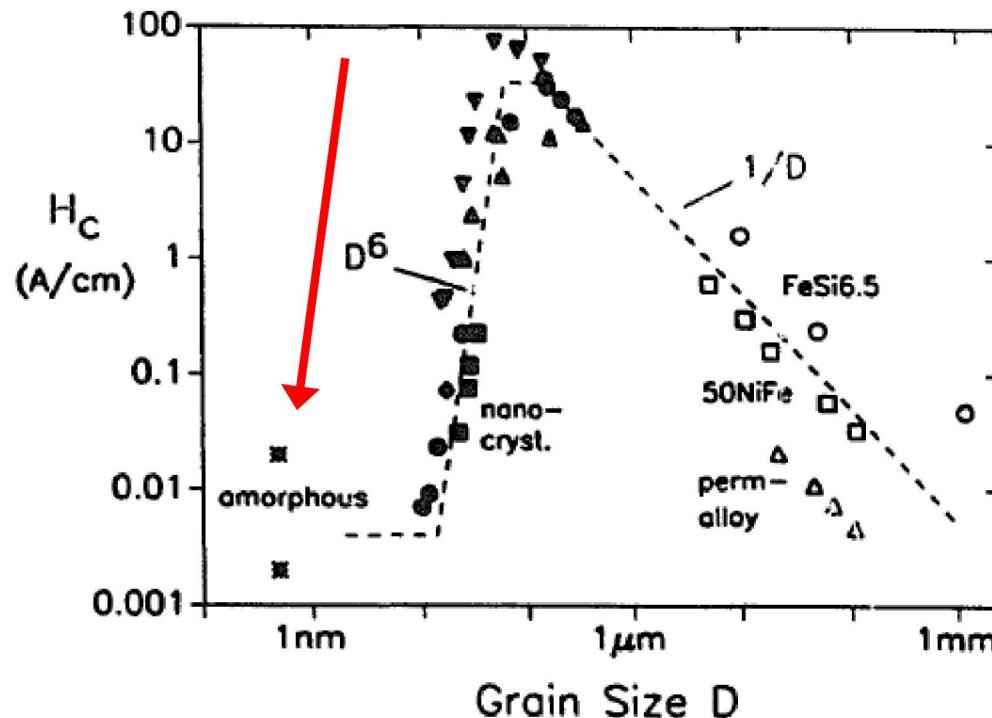
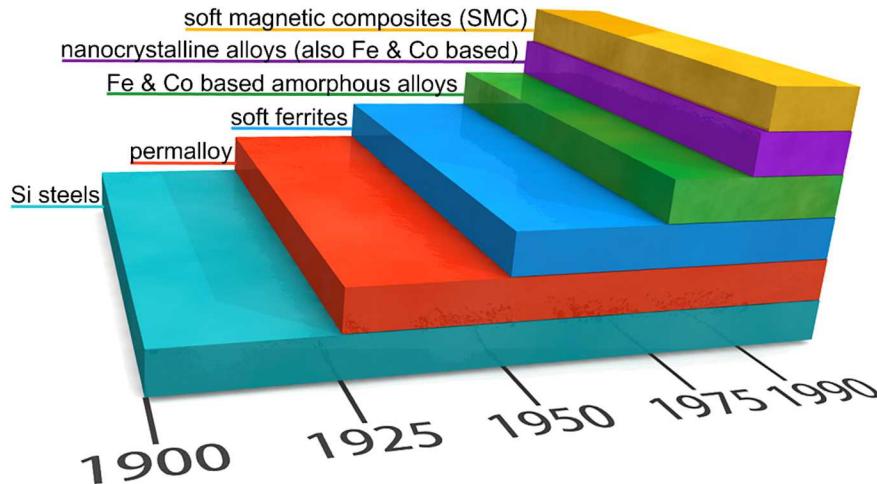


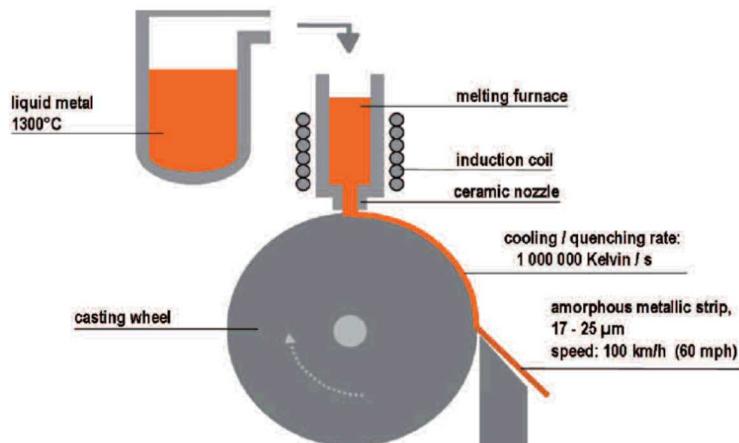
Fig. 2. Coercivity  $H_c$  vs. grain size for various soft magnetic metallic alloys. The data of the nanocrystalline material refer to (▲) FeNbSiB and (●) FeCuNbSiB [14], (◆) FeCuVSiB [15], (■) FeZrB [4] and (▼) FeCoZr [16].

G. Herzer, Nanocrystalline Soft Magnetic Materials, J. Magn. Mag. Mat., 112, 258 (1992).

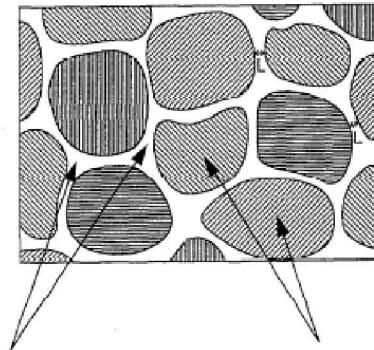
# Development of Soft Magnetic Materials



Adapted from: L.A. Dobrzański, M. Drak, B. Ziębowicz, Materials with specific magnetic properties, Journal of Achievements in Materials and Manufacturing Eng., 17, 37 (2006).



VITROPERM  
(Vacuumschmelze)



Intergranular amorphous phase with high  $T_c$  and high thermal stability due to large amounts of M and B elements.

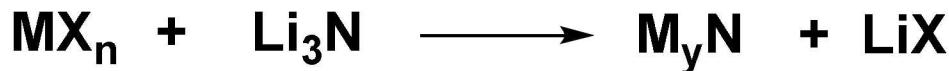
Nano-scale  $\alpha$ -Fe grains with small  $\lambda$  due to small amounts of M and B elements.

## “NANOPERM”

A. Makino, et. al., Nanocrystalline Soft Magnetic Fe-M-B (M = Zr, Hf, Nb) alloys and their applications, Mat. Sci. and Eng., A226-228, 594 (1997).

- Complex stoichiometry including Fe, Co, and other inactive elements such as B, Zr, Hf, Nb, Cu, Mo, Si, C
- Time consuming and high temperature processing → costly!
- Some inactive material to form a low loss nanocrystalline structure

# Metathesis- effective for early transition metals



- Potentially facile route to iron nitride

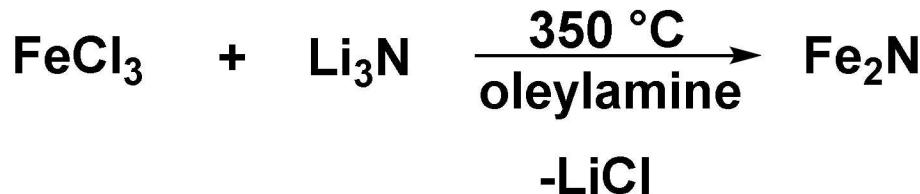
Group	MX <sub>n</sub>	Product
4	TiCl <sub>4</sub>	TiN
	ZrCl <sub>4</sub>	ZrN
	HfCl <sub>4</sub>	HfN
5	VCl <sub>3</sub>	VN, V <sub>2</sub> N
6	CrCl <sub>3</sub>	Cr, Cr <sub>2</sub> N
7	MnI <sub>2</sub>	Mn <sub>4</sub> N, Mn
8	FeCl <sub>3</sub>	Fe
10	NiCl <sub>2</sub>	Ni

- Effective for the early transition metals

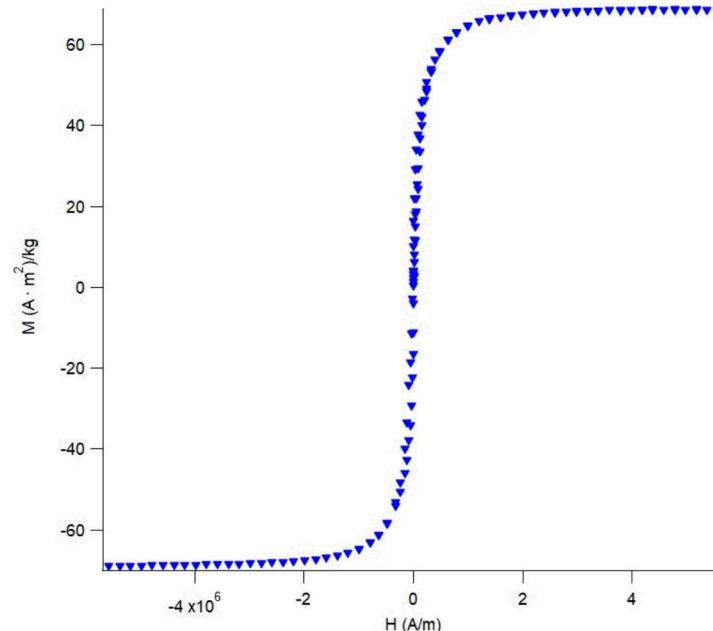
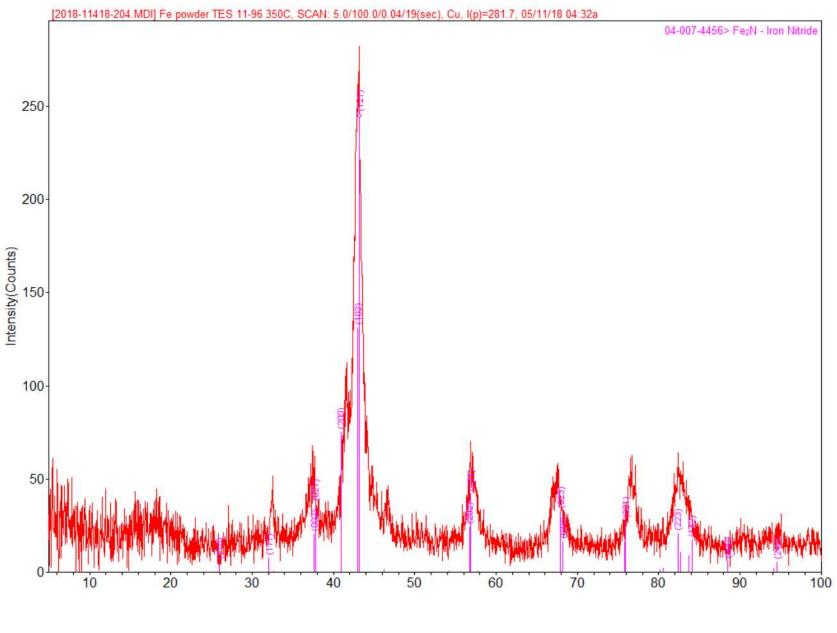


- Reactions with late metals generate too much heat
  - Decomposition to elemental metal

# Slow addition of $\text{FeCl}_3$ results in phase pure $\text{Fe}_2\text{N}$



- 0.124 M  $\text{FeCl}_3$  oleylamine
- Injected at 80  $\mu\text{L}/\text{min}$
- 4 hour reaction time



# Fe<sub>2</sub>N is generally not observed in isolation



Science and Technology of Advanced Materials 5 (2004) 83–87

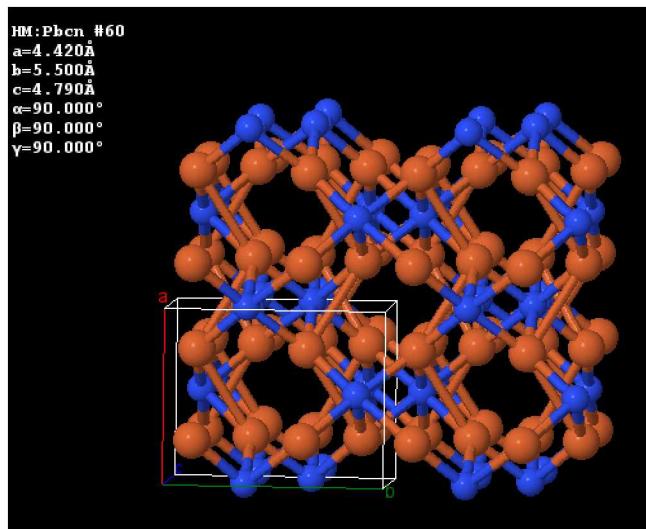
SCIENCE AND  
TECHNOLOGY OF  
ADVANCED  
MATERIALS  
[www.elsevier.com/locate/stam](http://www.elsevier.com/locate/stam)

## Magnetic properties of weak itinerant ferromagnetic $\zeta$ -Fe<sub>2</sub>N film

Hiroshi Naganuma<sup>a</sup>, Yasushi Endo<sup>a,b</sup>, Ryoichi Nakatani<sup>a,b,c</sup>,  
Yoshio Kawamura<sup>a</sup>, Masahiko Yamamoto<sup>a,b,\*</sup>



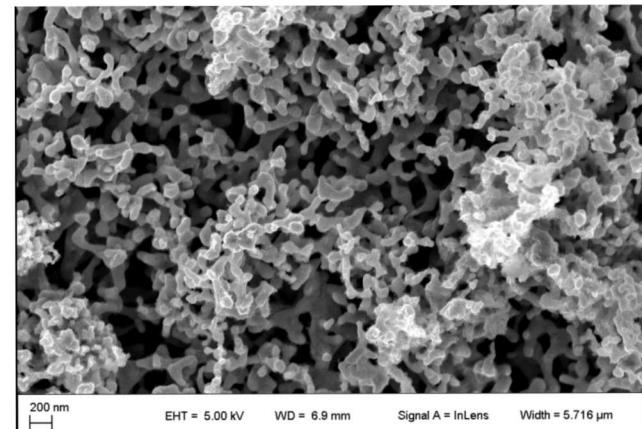
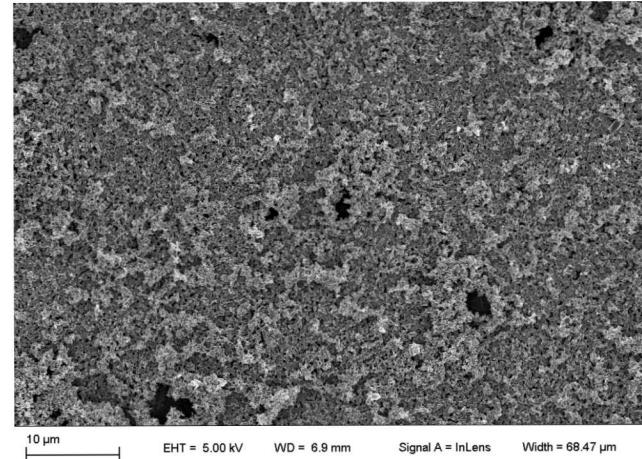
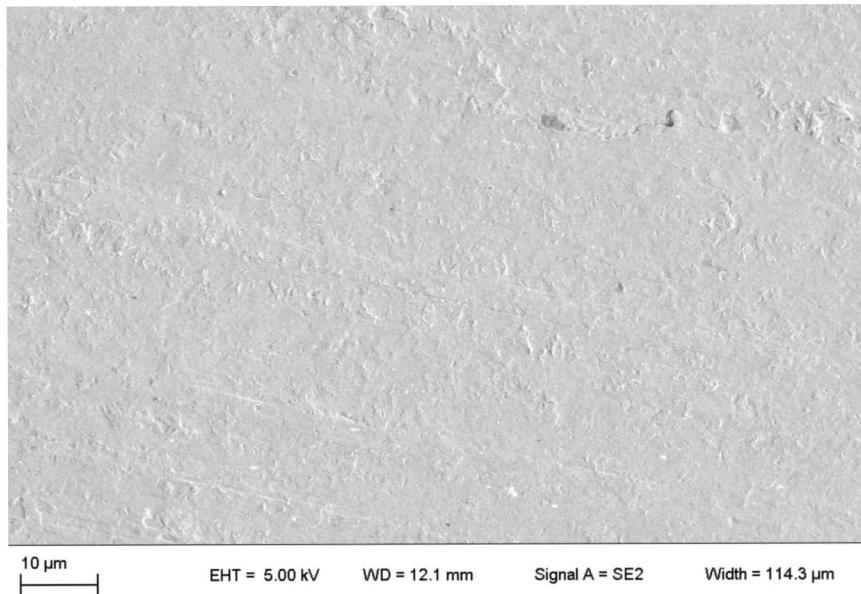
## Fabrication of $\epsilon$ -Fe<sub>2</sub>N Catalytic Sites in Porous Carbons Derived from an Iron–Triazolate Crystal



- Orthorhombic structure
- Further magnetic characterization is underway

# Electrochemical Nitridation of Fe(0)- LiCl/KCl

- Subsequent attempts have produced nitride layer



- A phase evolution vs. temperature (XRD) suggests FeN is the predominant phase

# Other Magnetic Nitrides of Interest

Material	Phase	$\sigma_s$ (Am <sup>2</sup> /kg)	$J_s$ (T), if available	$T_c$ (K)	$H_c$ (A/m)
FeN	rocksalt (fcc or fct)	209			
$\gamma'$ -Fe <sub>4</sub> N	antiperovskite-like	209	1.89	769	460
$\alpha''$ -Fe <sub>16</sub> N <sub>2</sub>	tetragonal	230 - 286	2.3	810	
$\alpha''$ -Fe <sub>90</sub> N <sub>10</sub>		230			
g-C <sub>4</sub> N <sub>3</sub>	graphitic	62			
MnN	rocksalt	194-308			4000
$\alpha$ -Fe	bcc	217	2.15	1044	70

**Nitrides will have higher resistivities than soft magnetic materials and will not require laminations of inactive material to mitigate eddy current losses**