

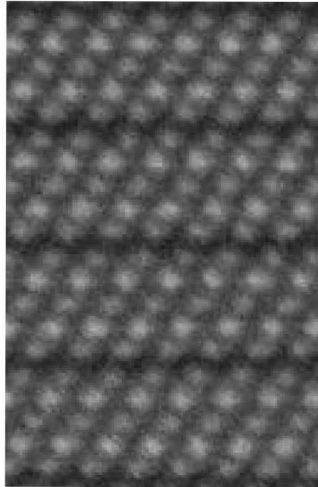
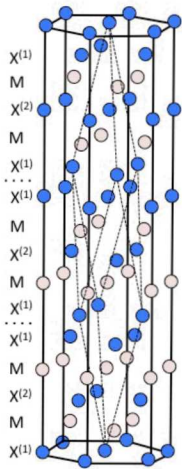
Extended Crystal Defects in Layered Chalcogenide Materials: Atomic-scale Structural Investigations

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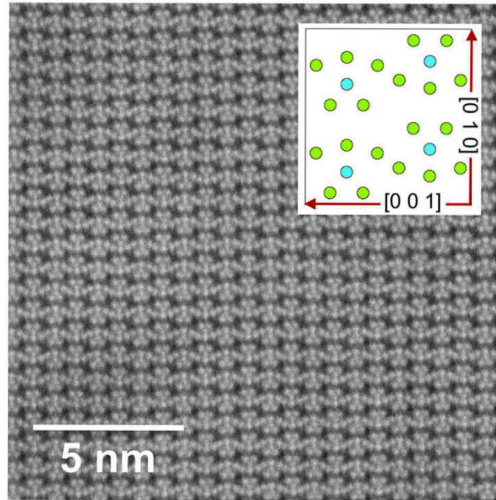
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Layered Chalcogenides: Diversity of compositional and structural arrangements

Example: Bi_2Te_3



Example: ZrTe_5



Anisotropic, layered structures contribute to useful and interesting thermal, electronic, optical properties:

***Thermoelectrics,
Photothermoelectric Effect,
Topological Insulators,
Dirac Semimetals***

Focus for today's presentation:

Analysis of dislocation core-structure in these materials

Bismuth Telluride:

- Basal dislocations – core spreading in relationship to gamma surface
- Rhombohedral dislocation – climb-dissociation

Zirconium Telluride:

- [001] dislocation – climb dissociation

Collaborators

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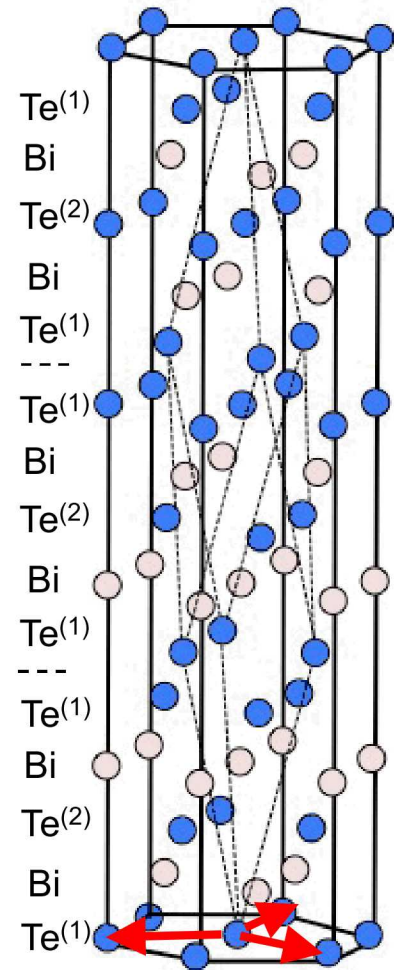
L. Hale

LBNL/National Center for Electron Microscopy

C. Ophus

Dislocations in Bismuth Telluride (Bi_2Te_3)

Burgers vectors lying in basal plane



Array of $\frac{1}{3}\langle 2-1-1\ 0 \rangle$ Dislocations in Bi_2Te_3

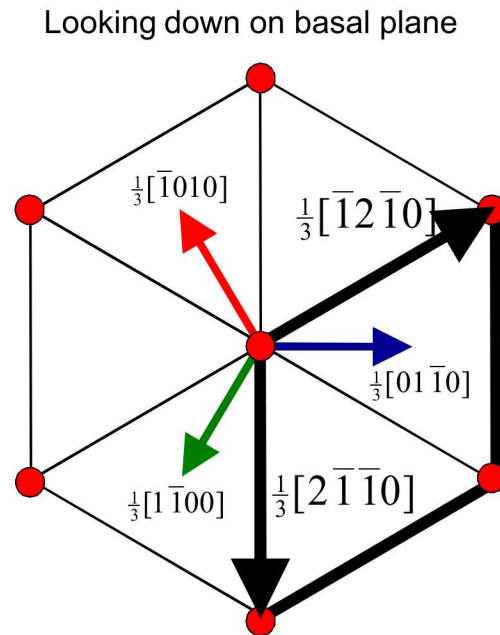
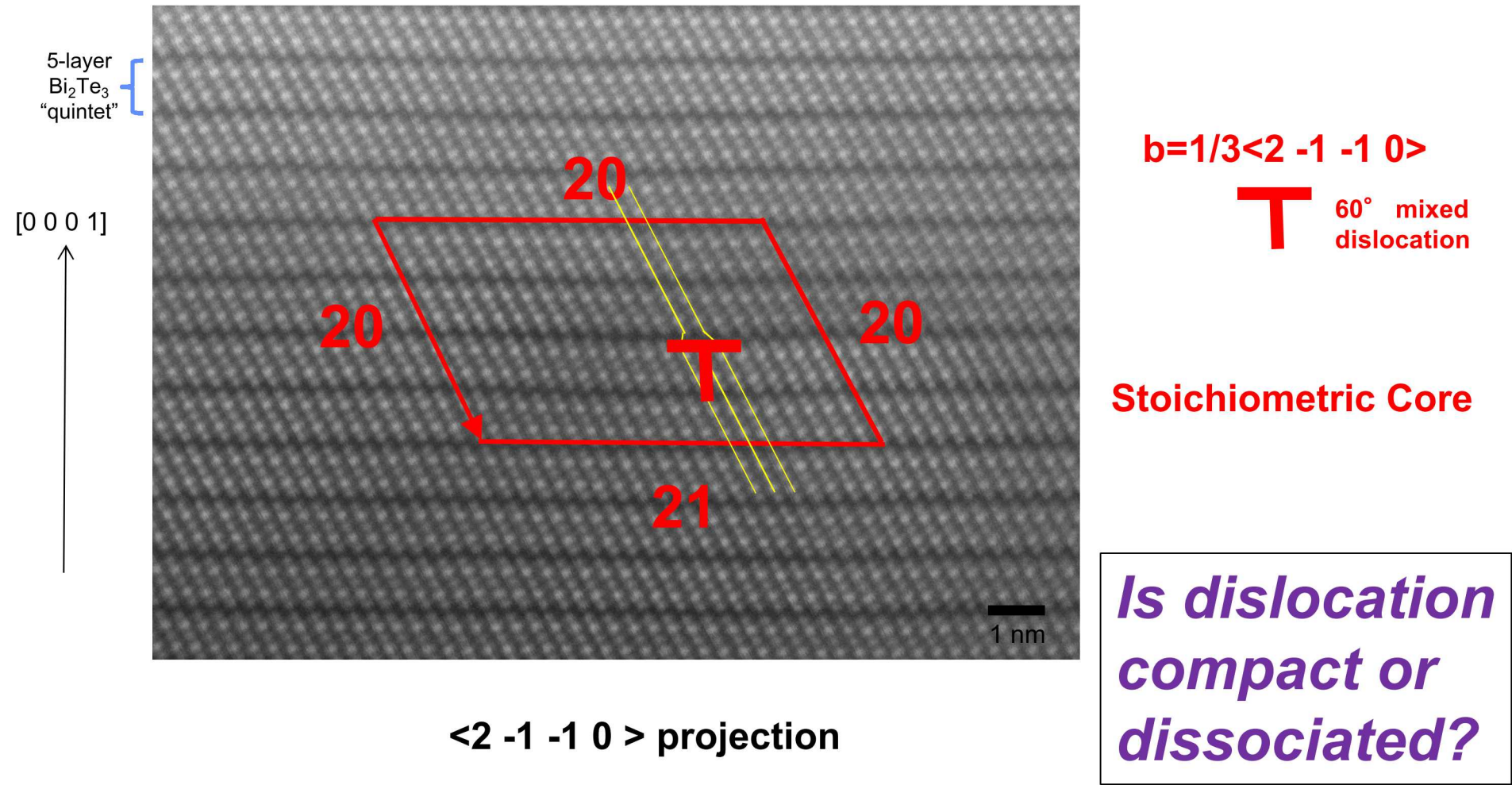


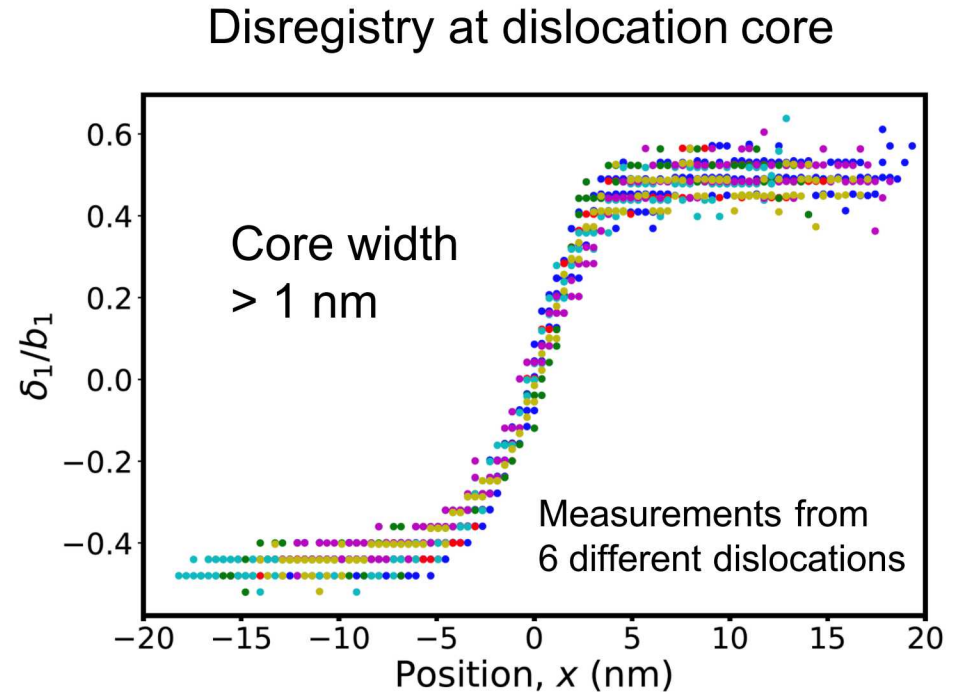
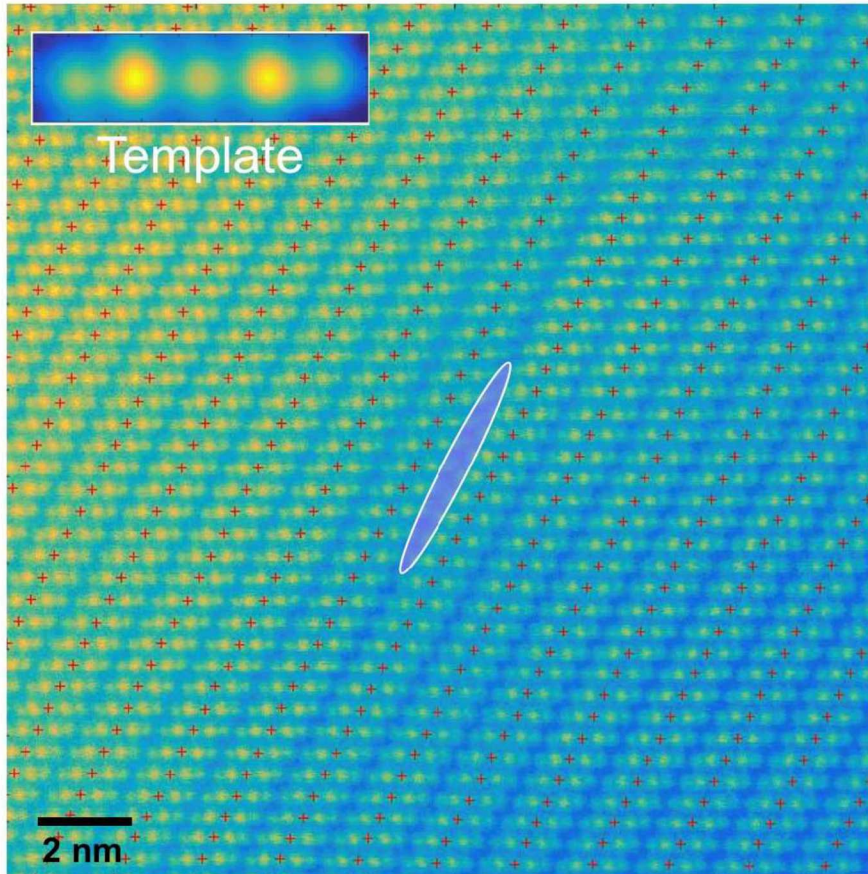
Fig. 1. Dislocation network in Bi_2Te_3 . Note that certain segments of dislocations and certain node-points have left the foil

Weak van der Waals bonding between tellurium planes

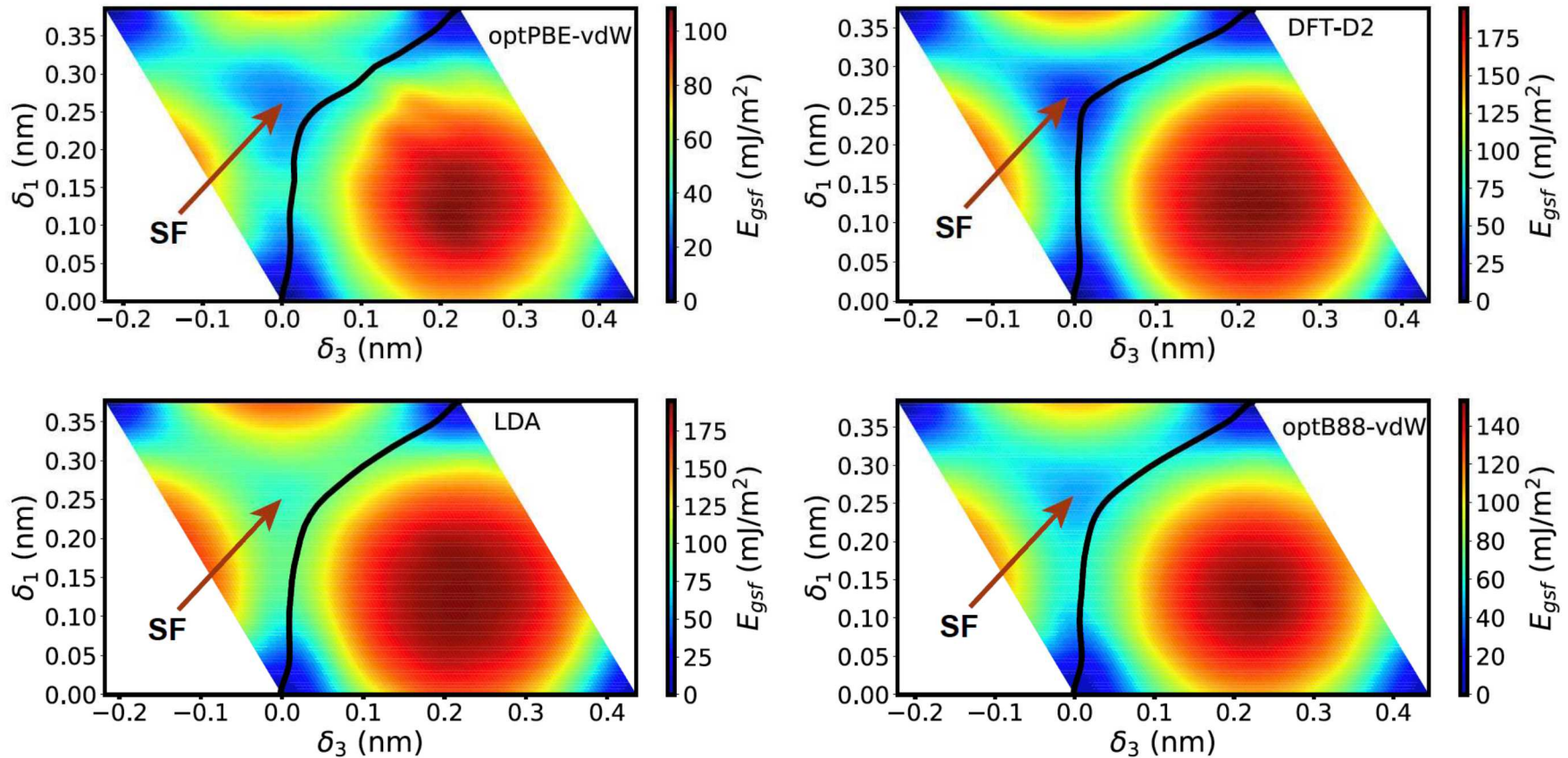
Dislocation Core structure: Termination at $\text{Te}^{(1)}\text{-Te}^{(1)}$ layer



Surprising Result: Broad core, but no dissociation into partial dislocations



Modeling: combination of ab initio and semi-continuum dislocation theory

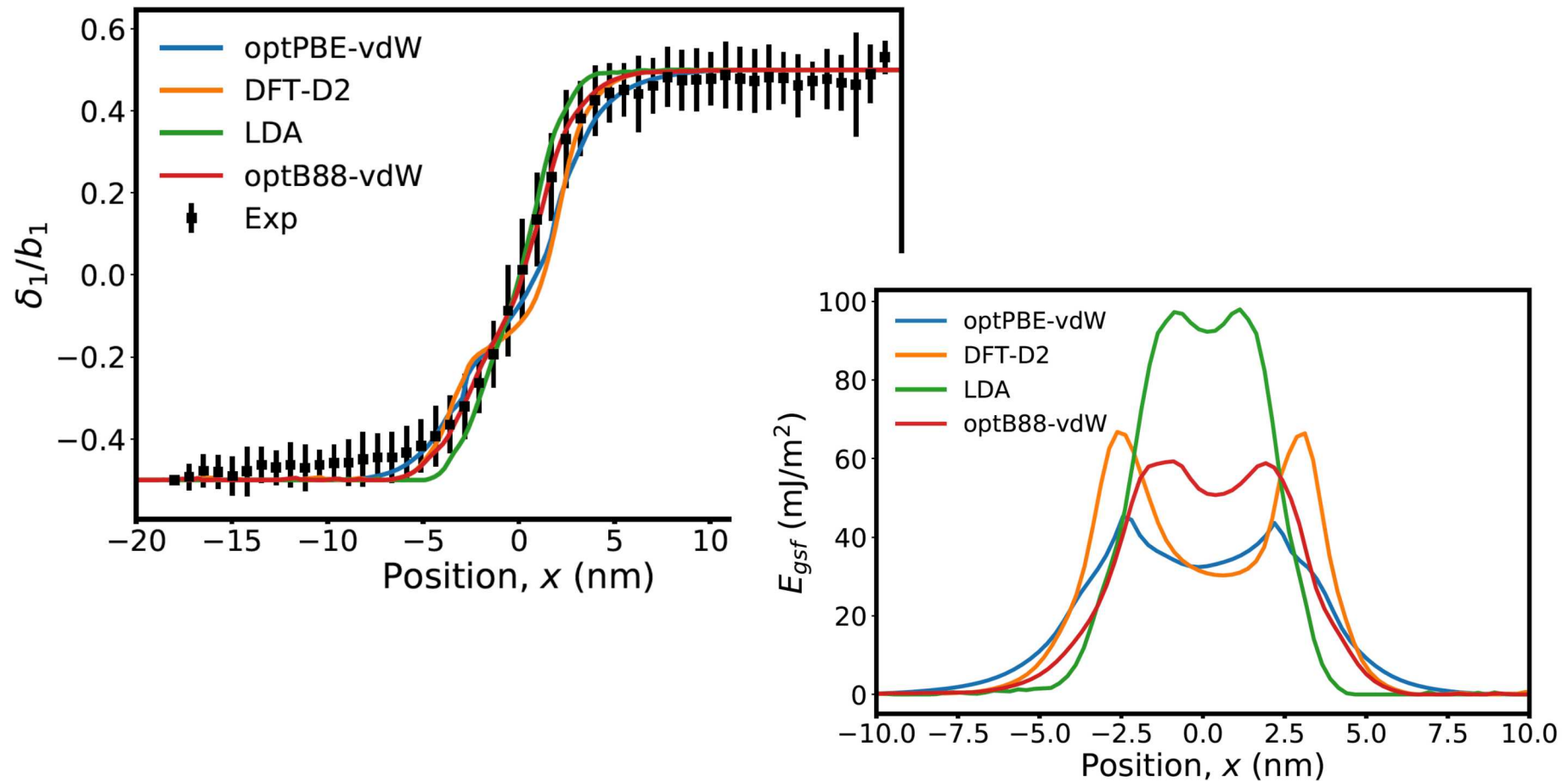


DFT calculations for generalized SFE using different van der Waals corrections

Peierls-Nabarro model for dislocation disregistry:

-accounts for anisotropic elastic parameters of Bi_2Te_3

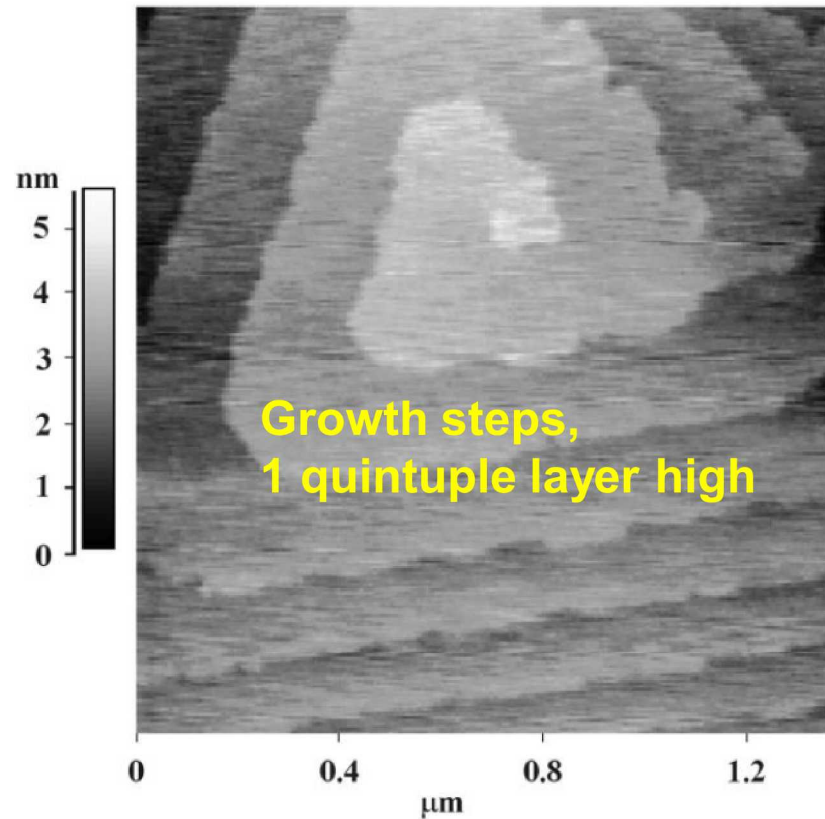
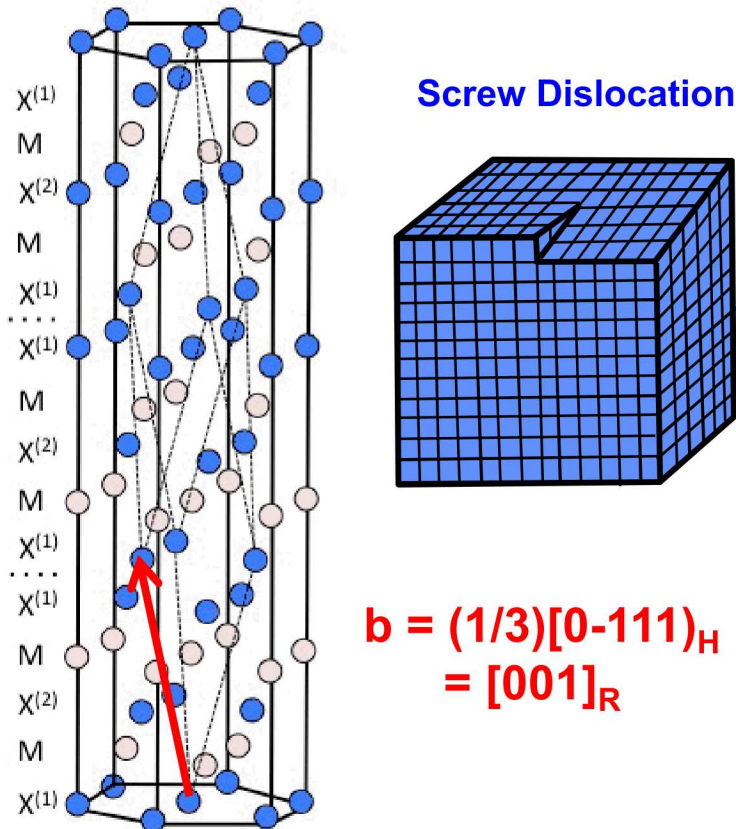
Shallow minimum relative to unstable SFE prevents localization into discrete partials



Non-basal dislocations:

Screw dislocations important to crystal growth

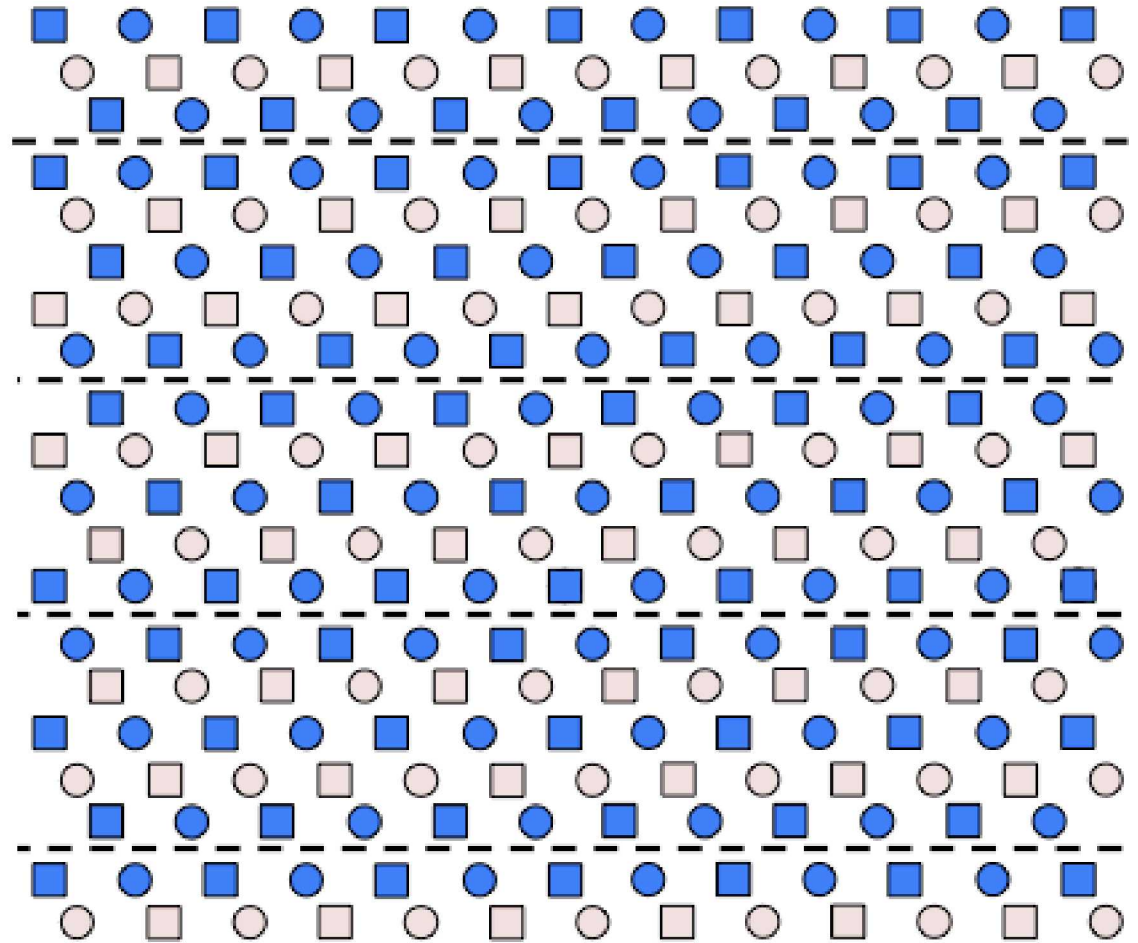
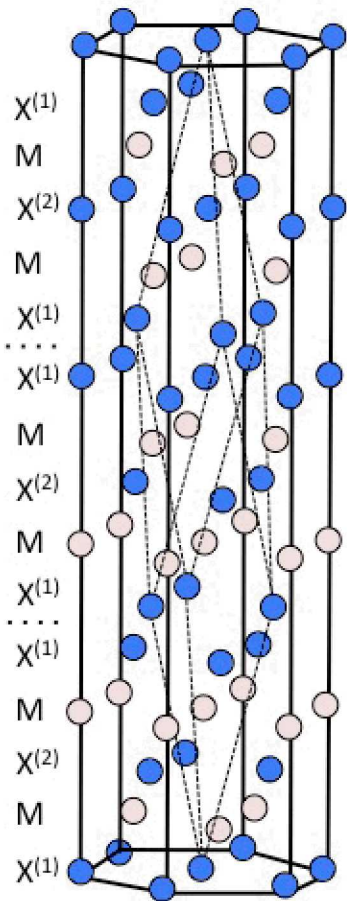
Example: spiral growth steps at screw dislocation in Bi_2Te_3 thin film



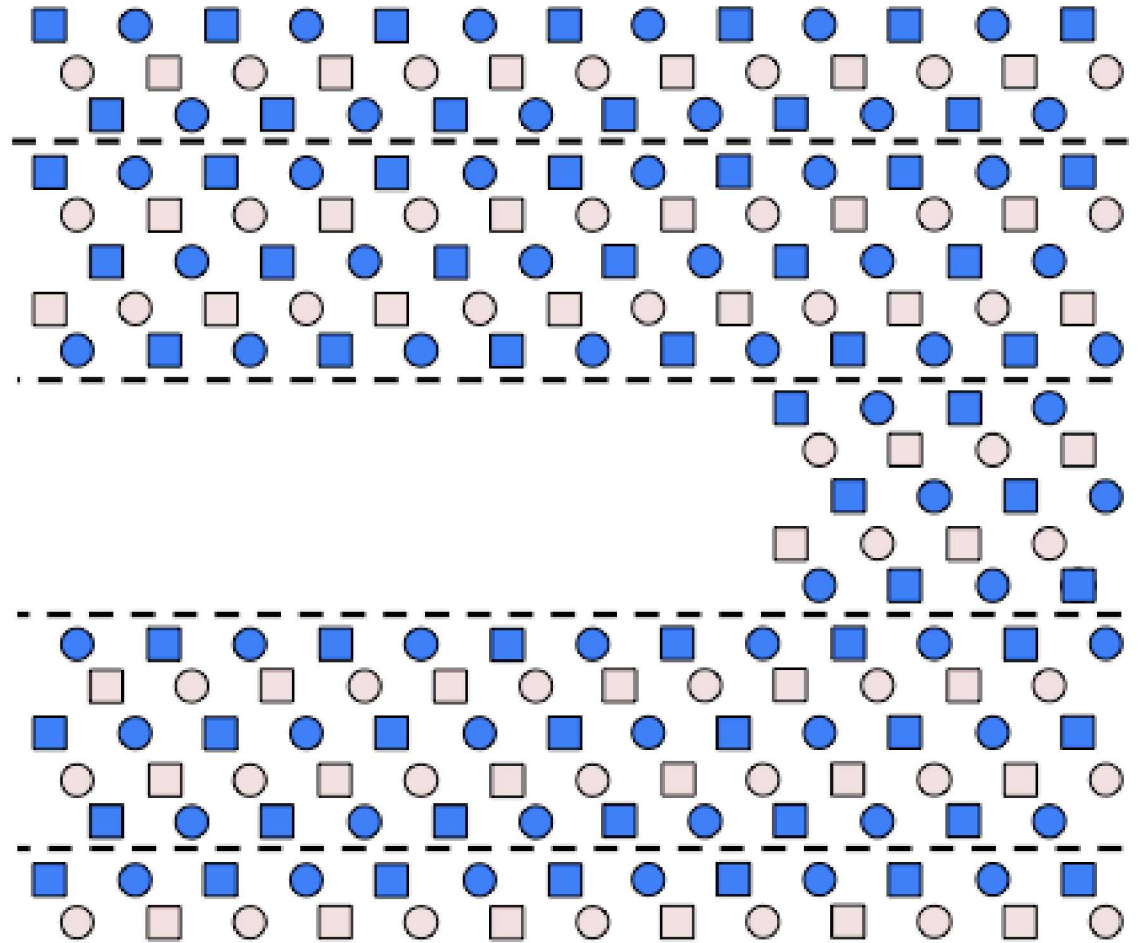
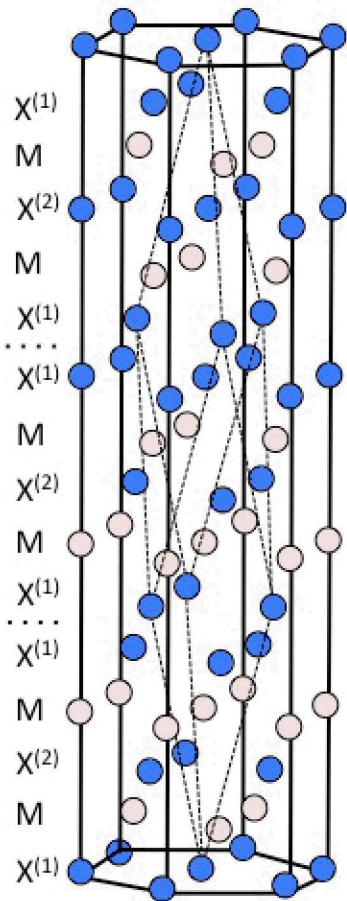
What about **edge** dislocations
with non-basal Burgers vectors?

M. Ferhat, J.C. Tedenac, J. Nagao,
J. Crystal Growth (2000)

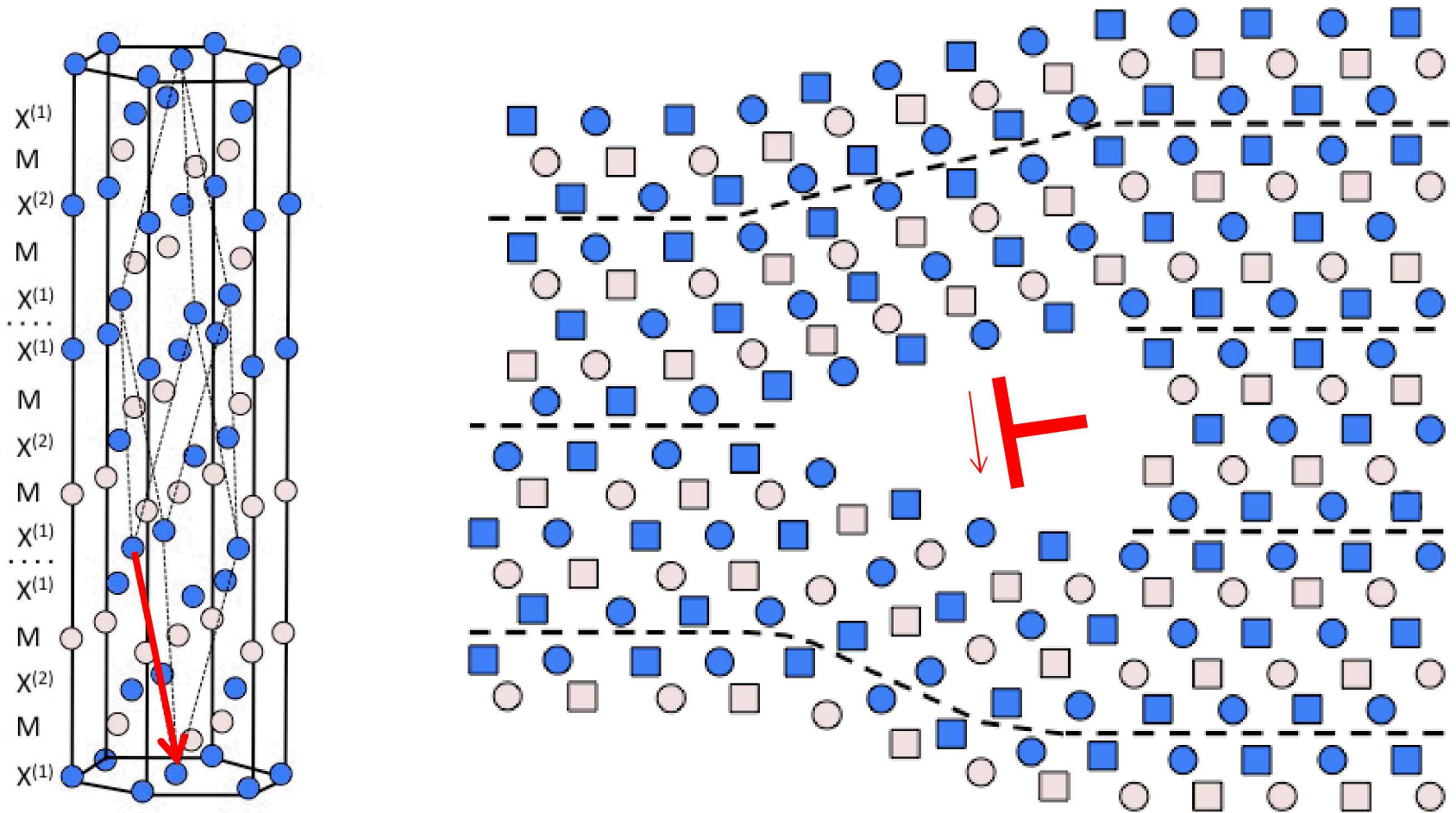
Non-basal **edge** dislocations: What happens if we pull out a quintuple unit?



Non-basal **edge** dislocations: What happens if we pull out a quintuple unit?

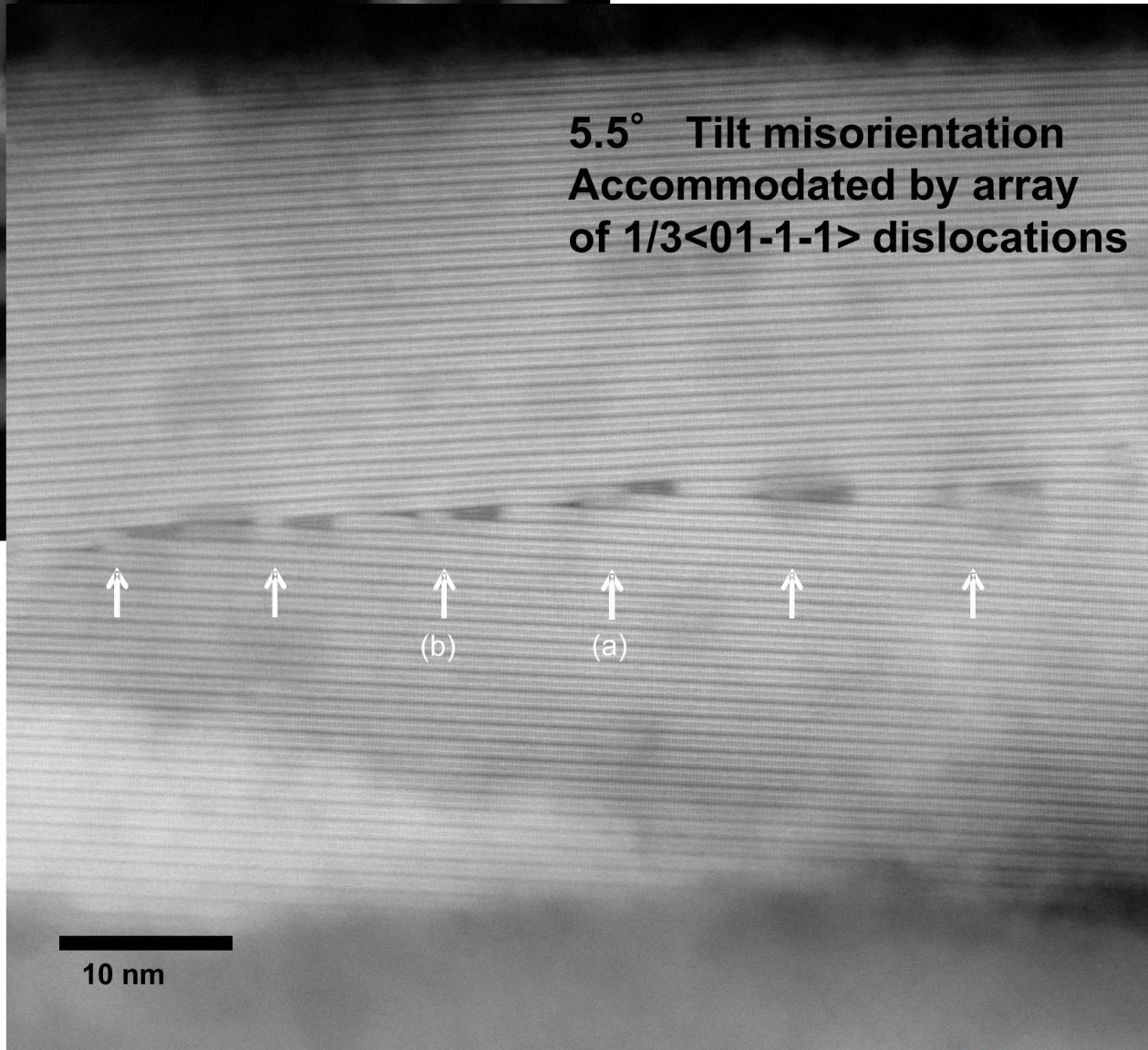
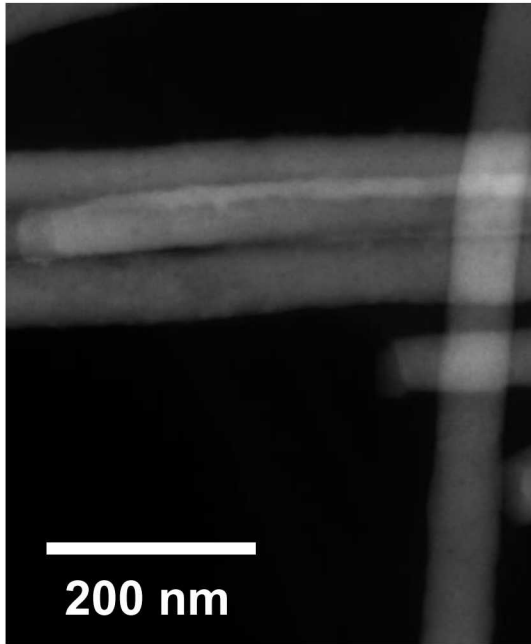


Non-basal **edge** dislocations: What happens if we pull out a quintuple unit?



$$\mathbf{b} = (1/3)[01\bar{1}\bar{1}]_H = [00\bar{1}]_R$$

Dislocations in Bi_2Te_3 Nanowires

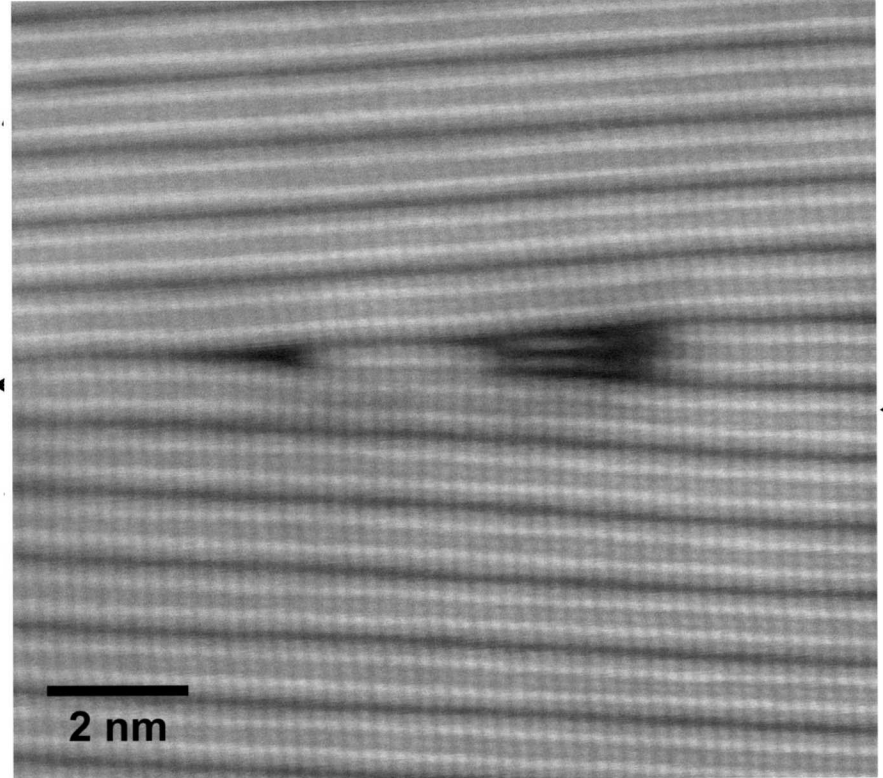
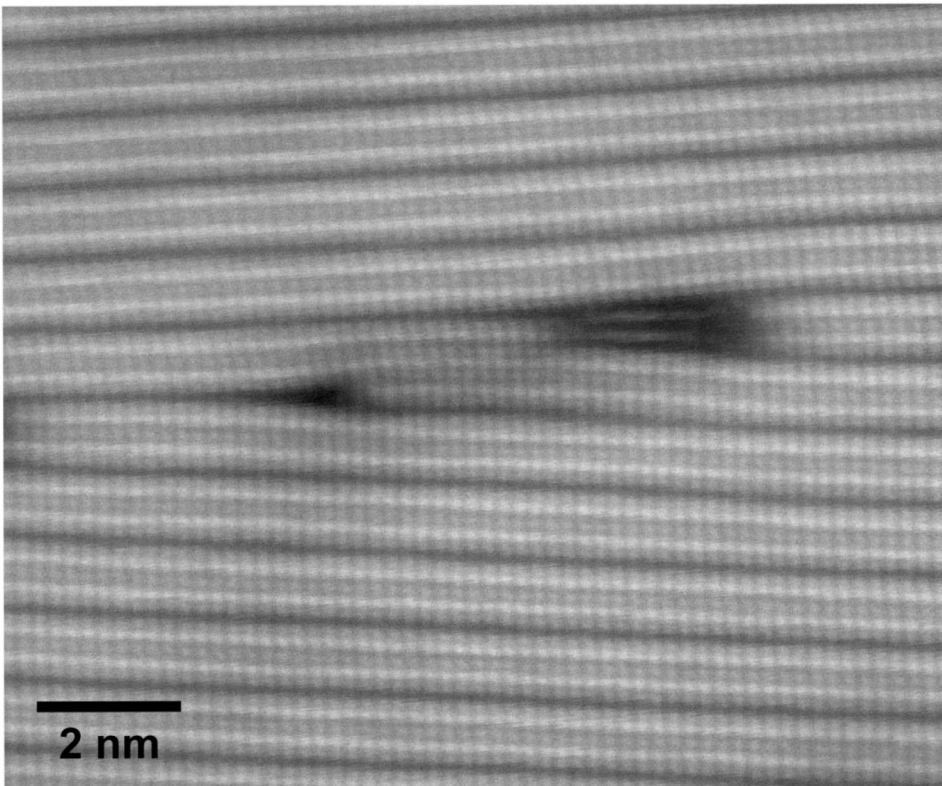


Wires formed by electrochemical deposition in nanoporous AAO templates.

Free standing wires annealed 30 minutes at 300° C in Ar-3% H_2 .

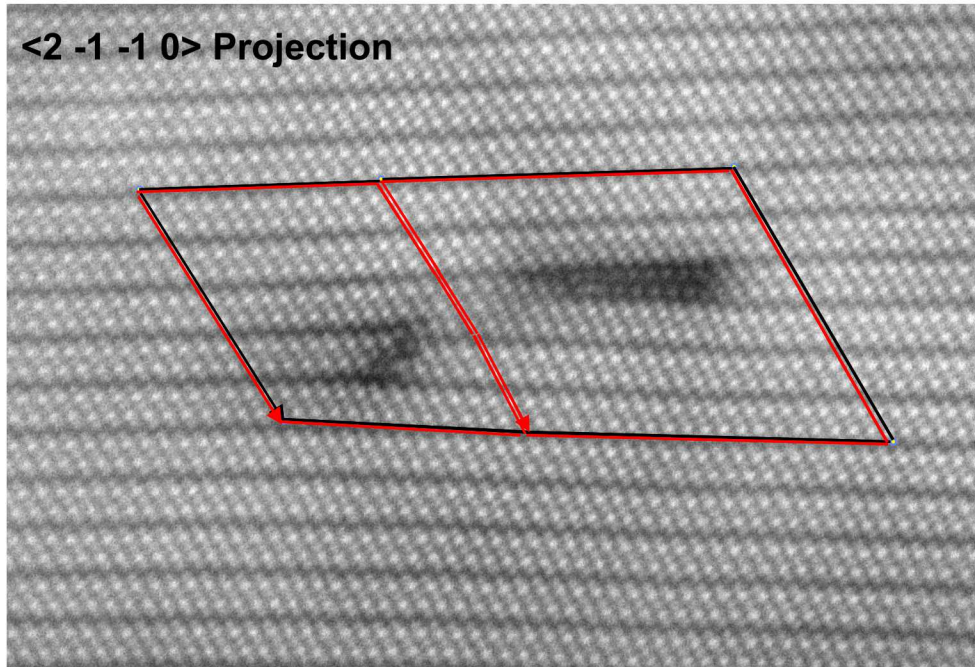
→ *Some loss of Te due to high vapor pressure*

Dislocations have dissociated core: two configurations

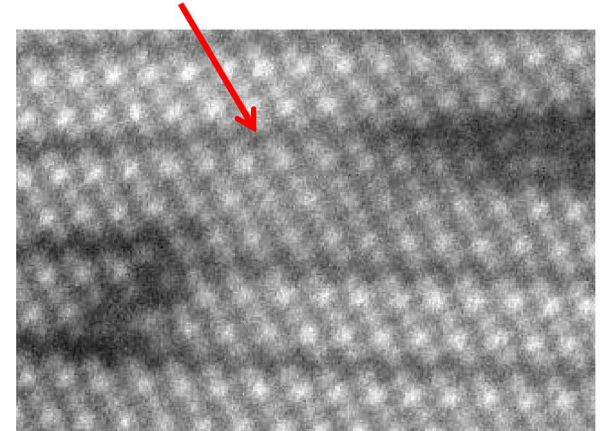


$\langle 1\ 0\ \bar{1}\ 0 \rangle$ Projection

$(1/3)[0\ 1\ -1\ -1]$ Dislocation in Bi_2Te_3 :



Core structure:
 Bi_3Te_4 7-layer fault



$$\mathbf{b} = (1/3) [0\ 1\ -1\ -1] \rightarrow (1/15) [0\ 5\ -5\ -2] + (3/15) [0\ 0\ 0\ -1]$$

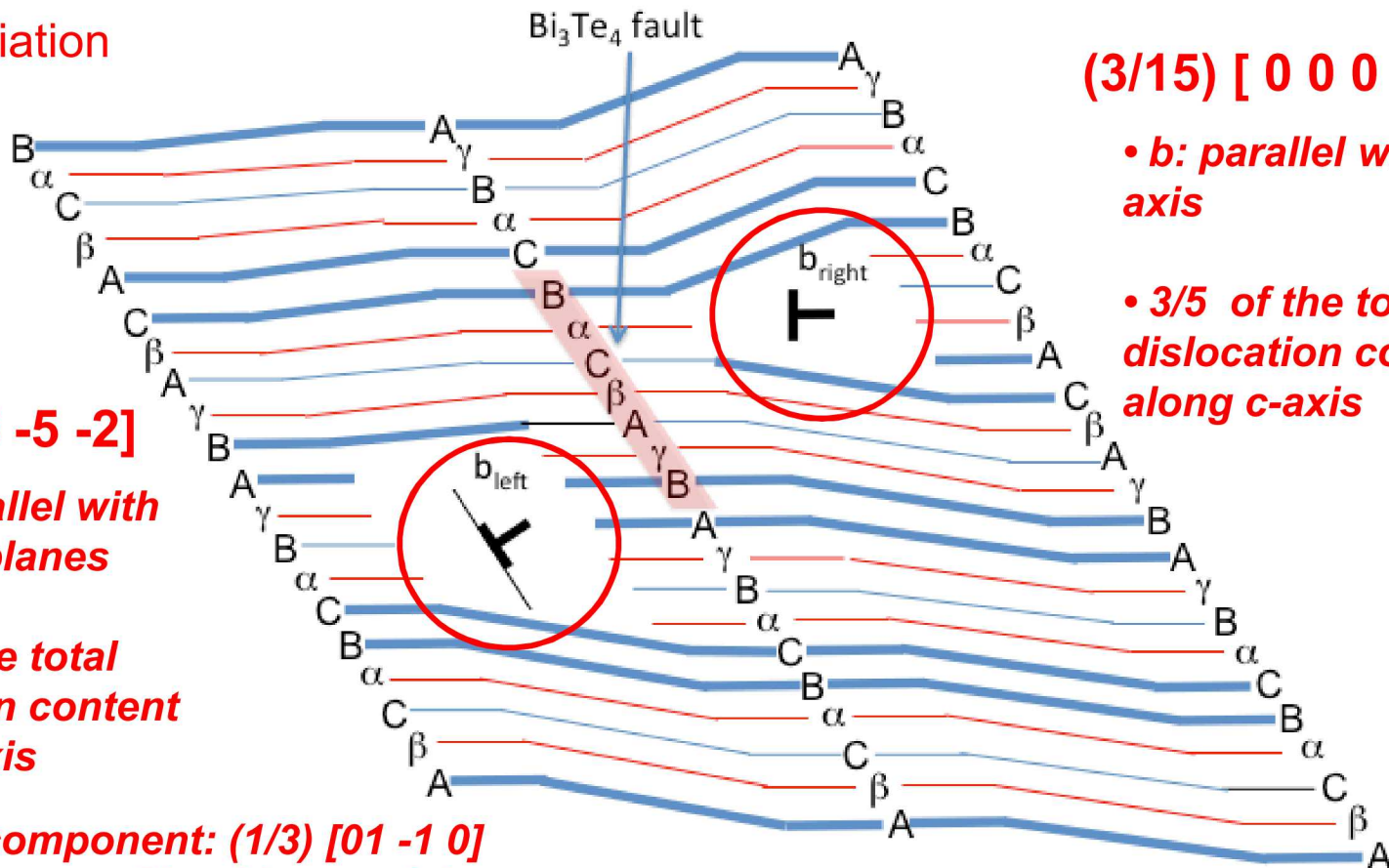
$$b^2 > b_1^2 + b_2^2$$

$$109.8\ \text{\AA}^2 > 22.9\ \text{\AA}^2 + 37.2\ \text{\AA}^2 = 60.2\ \text{\AA}^2$$

Reduced strain energy with dissociation

7-Layer Bi_3Te_4 faults: Mechanism to accommodate Te loss during annealing

Climb dissociation

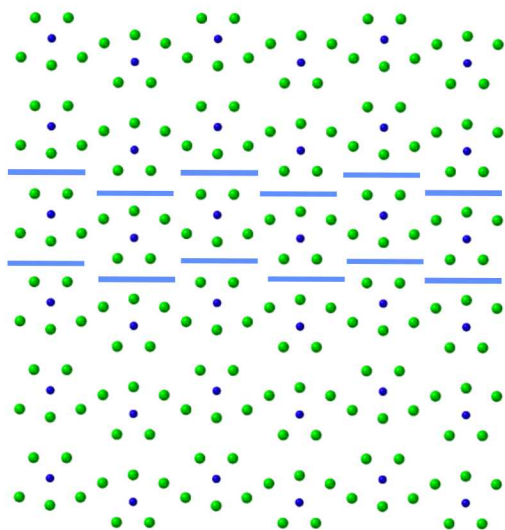


ZrTe₅: A complex, layered chalcogenide

Material of interest as a Dirac Semimetal

Corrugated layering structure

Orthorhombic, *Cmcm*

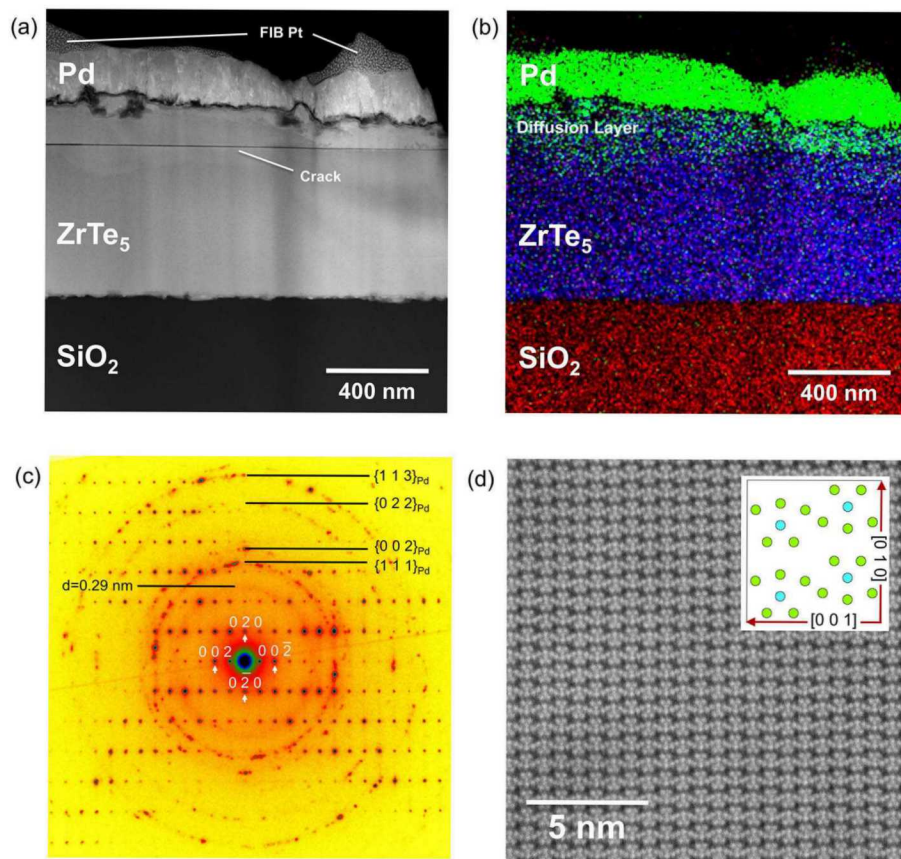


[1 0 0] projection

$a=3.9875\text{\AA}$, $b=14.530\text{\AA}$, $c=13.724\text{\AA}$

H. Fjellvåg, A. Kjekshus, Solid State Communications, 60 (2) (1986) 91-93

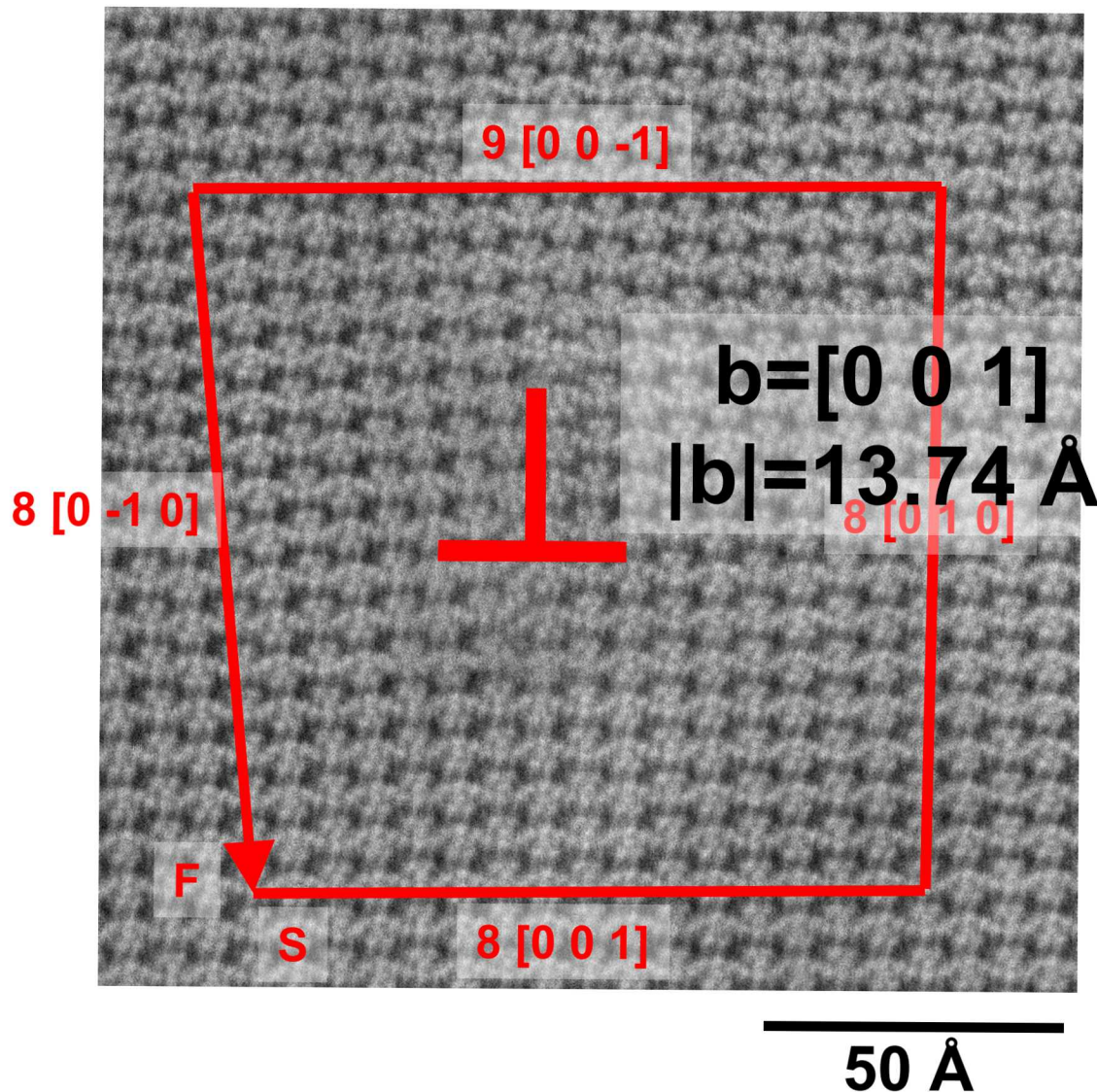
Interdiffusion of contact metallization causes large strains



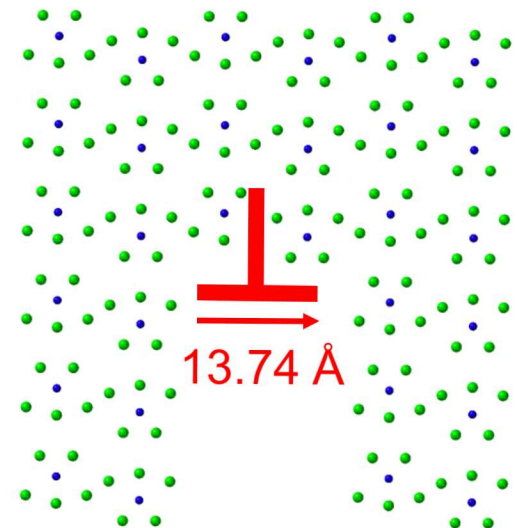
F. Léonard, W.L. Yu, K.C. Collins, D.L. Medlin, J.D. Sugar, A.A. Talin, W. Pan
ACS Applied Materials and Interfaces 9 (42) (2017) 37041-37047

Dislocation Structure in Zirconium Telluride

HAADF-STEM



ZrTe₅ Crystal structure
Orthorhombic, *Cmcm*



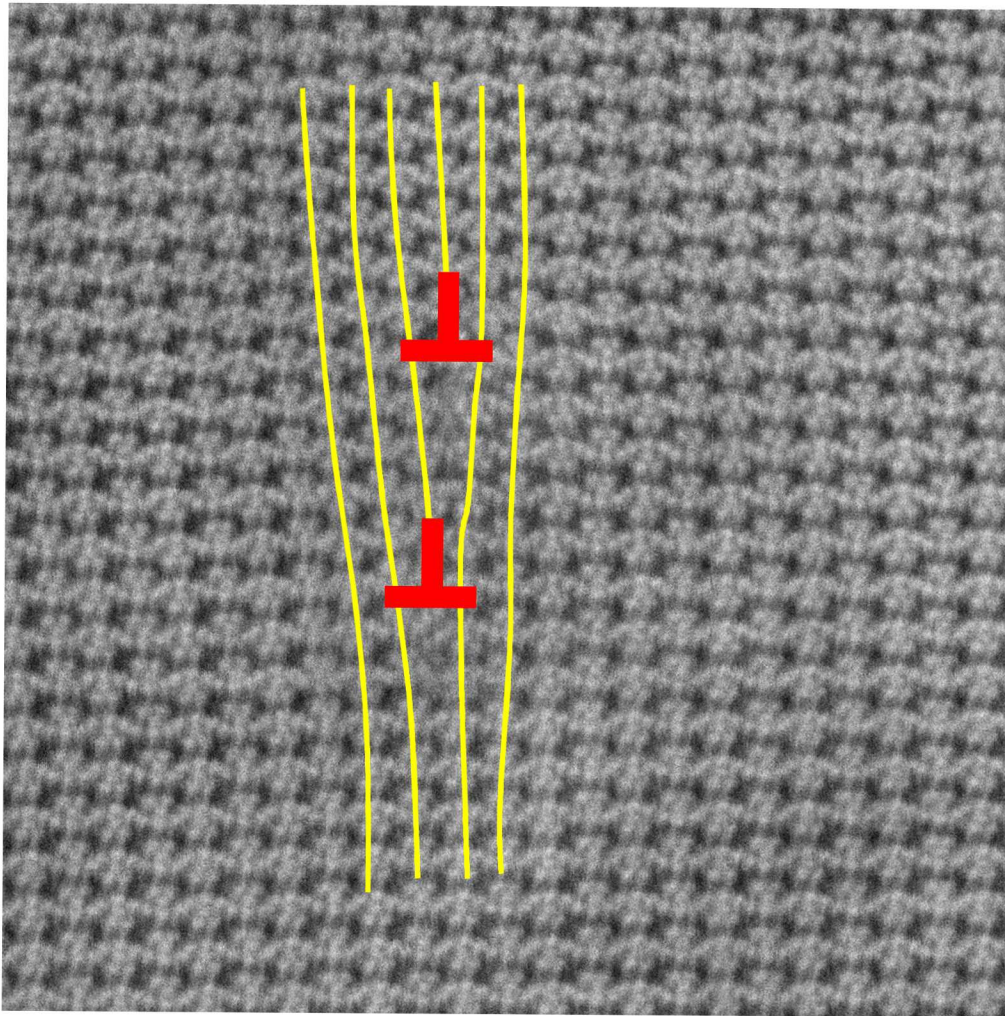
[1 0 0] projection

$a=3.9875\text{ Å}$, $b=14.530\text{ Å}$, $c=13.724\text{ Å}$

H. Fjellvåg, A. Kjekshus, Solid State Communications, 60 (2) (1986) 91-93

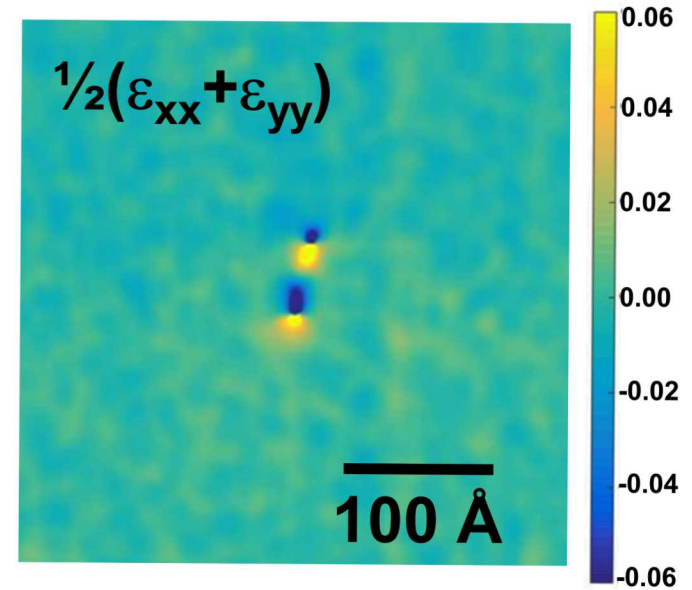
Dislocation Structure in Zirconium Telluride

HAADF-STEM



50 Å

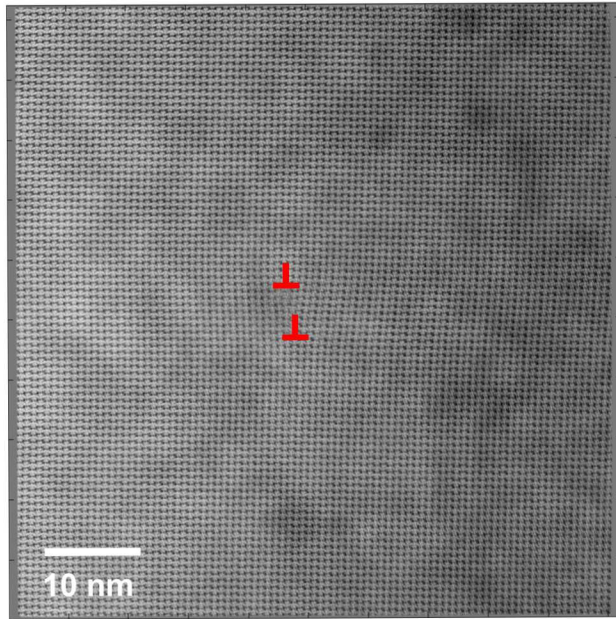
GPA Strain Map



**Climb dissociation
into two $\frac{1}{2}[0\ 0\ 1]$
partial dislocations**

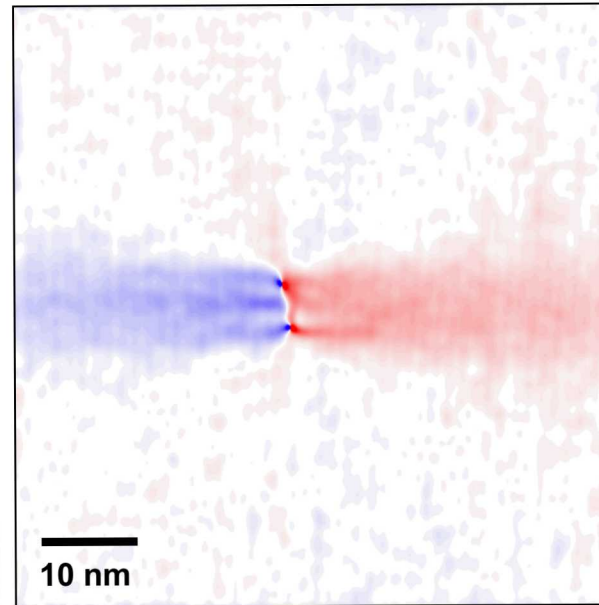
Elongation of shear strain along c-axis

HAADF-STEM

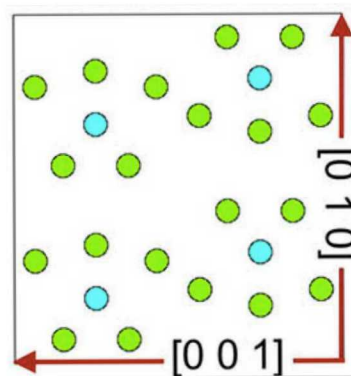
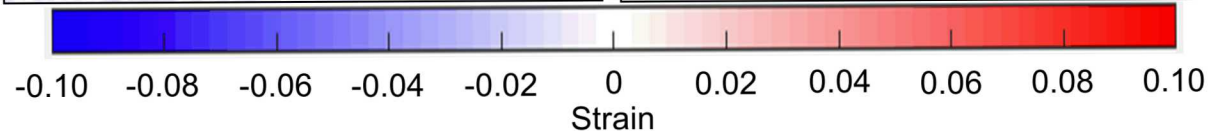
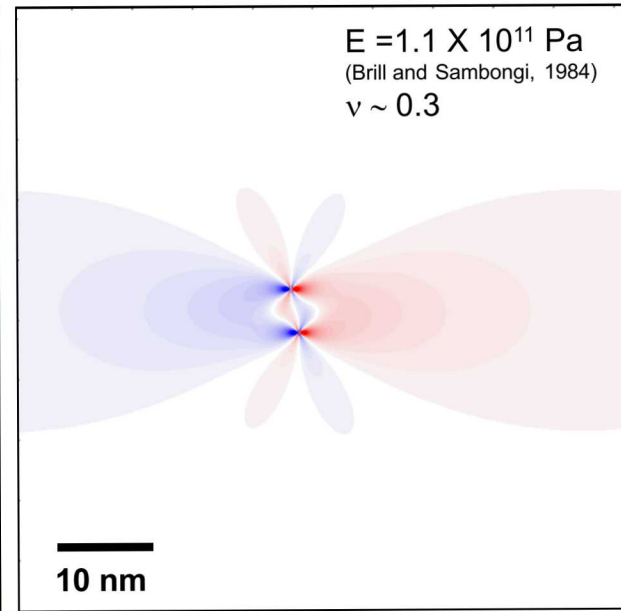


←
C-axis

ϵ_{xy} (measured, GPA)



ϵ_{xy} (calculated, isotropic elasticity)



←
c-axis

Measurement shows greater elongation of ϵ_{xy} region parallel to c-axis compared with isotropic elasticity solution.

Easier shear in direction parallel with double-tellurium layers?

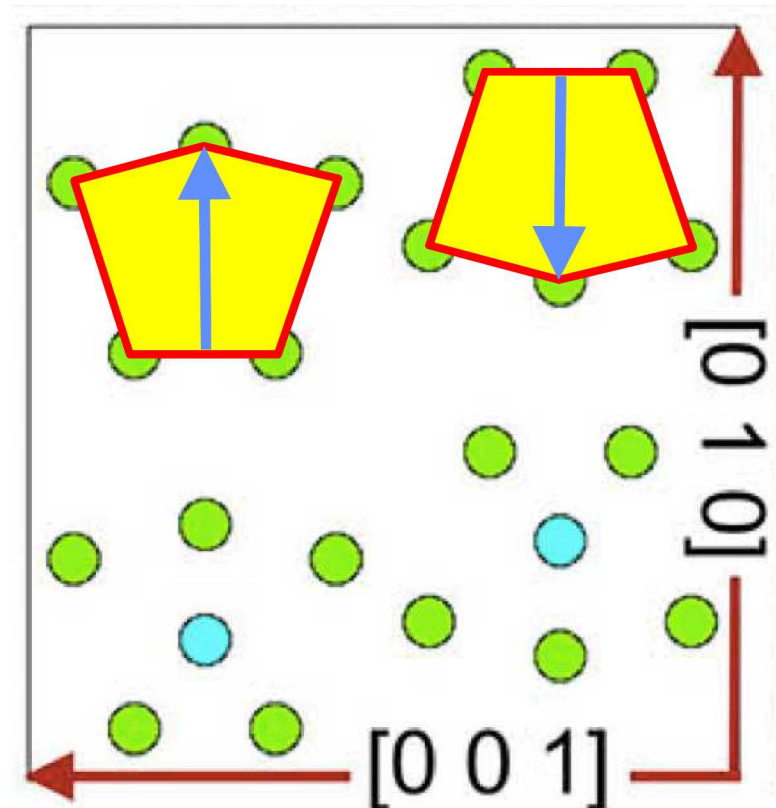
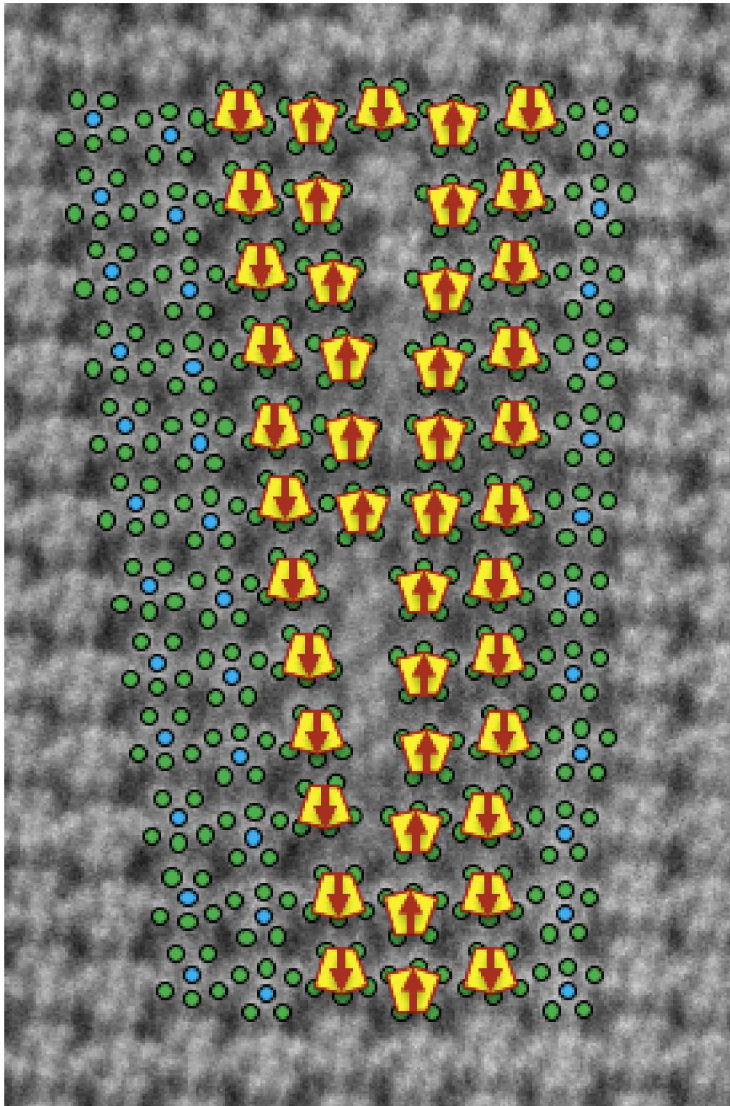
Image correction for non-linear scan distortion

C.J. Ophus, J. Ciston, C.T. Nelson, Ultramicroscopy (2016)

5 images collected at 90° scan rotation increments

GPA: Gaussian masks with $\sigma=0.12 \text{ nm}^{-1}$

Partial dislocations separate varying transition region: no well-defined fault



Conclusions

Key structural aspects layered chalcogenides are manifested in the detailed structures of extended defects in these materials.

Weak, van der Waals bonding across double chalcogenide layers

Ability to accommodate non-stoichiometry through altering the layer stacking

Atomic scale microscopy is clarifying the core structure of dislocations, interfaces, and interfacial line defects

Complex, dissociated core structures observed in Bi_2Te_3 and ZrTe_5

Theoretical treatments for transport behavior at such defects should not neglect the possibility for such dissociation.