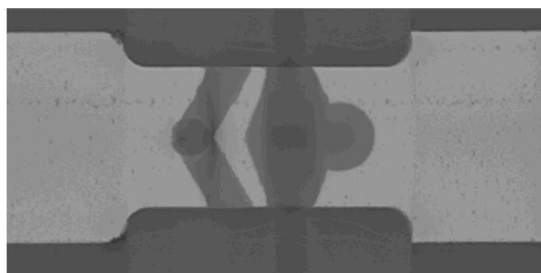
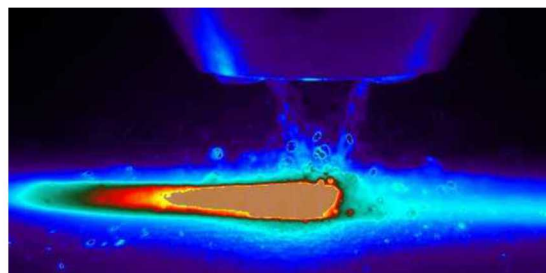


Multiscale Stochastic Modeling for Additive Manufacturing Part Qualification



PRESENTED BY

Kyle Johnson¹, John Emery¹, Mircea Grigoriu², Jay Carroll¹, Joe Bishop¹,
Kyle Karlson¹, Guy Bergel, Chris Hammetter¹, Spencer Grange¹, Kurtis
Ford¹, and Judy Brown¹

Engineering Mechanics Institute Conference 2019
June 20, 2019

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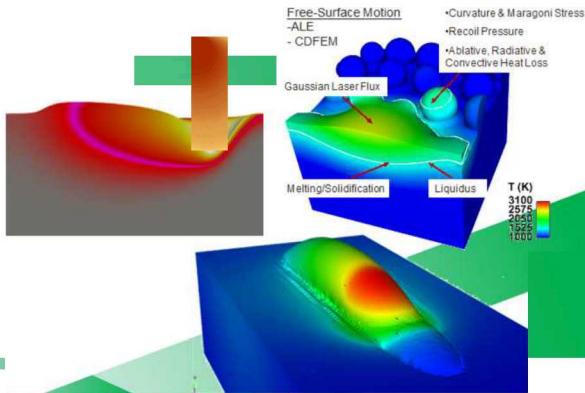
Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

2 Models Bridging Length Scales

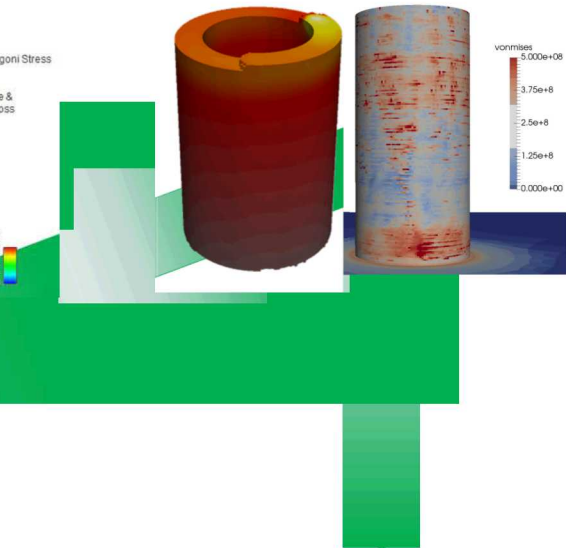


LAMMPS
ARIA
ADAGIO
SPPARKS

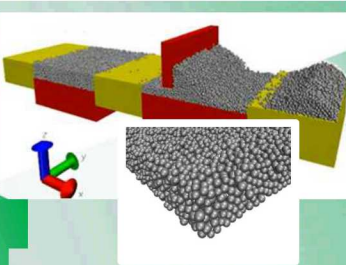
Solidification Scale Thermal
M. Martinez, B. Trembacki, D. Moser



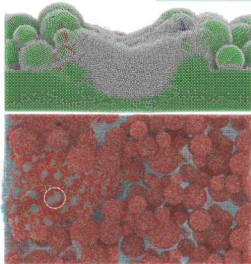
Build Scale Thermal + Mechanics
K. Johnson, K. Ford, L. Beghini, M. Stender & J. Bishop



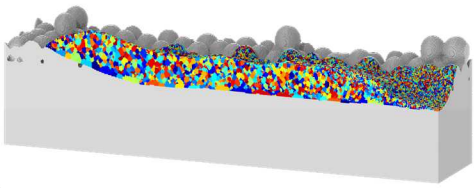
Powder Spreading
D. Bolintineanu



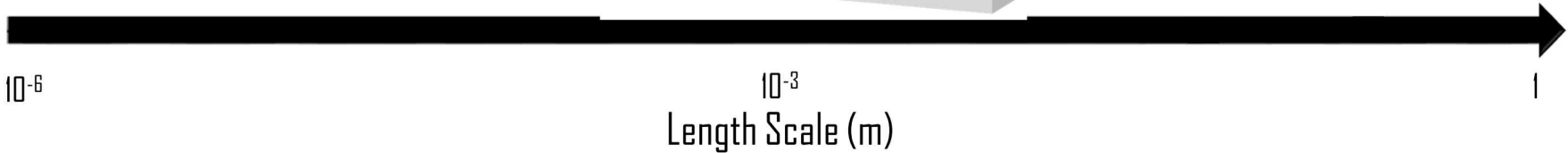
Powder Behavior
M. Wilson



Mesoscale Texture/Solid Mechanics
T. Rodgers, J. Brown, K. Ford

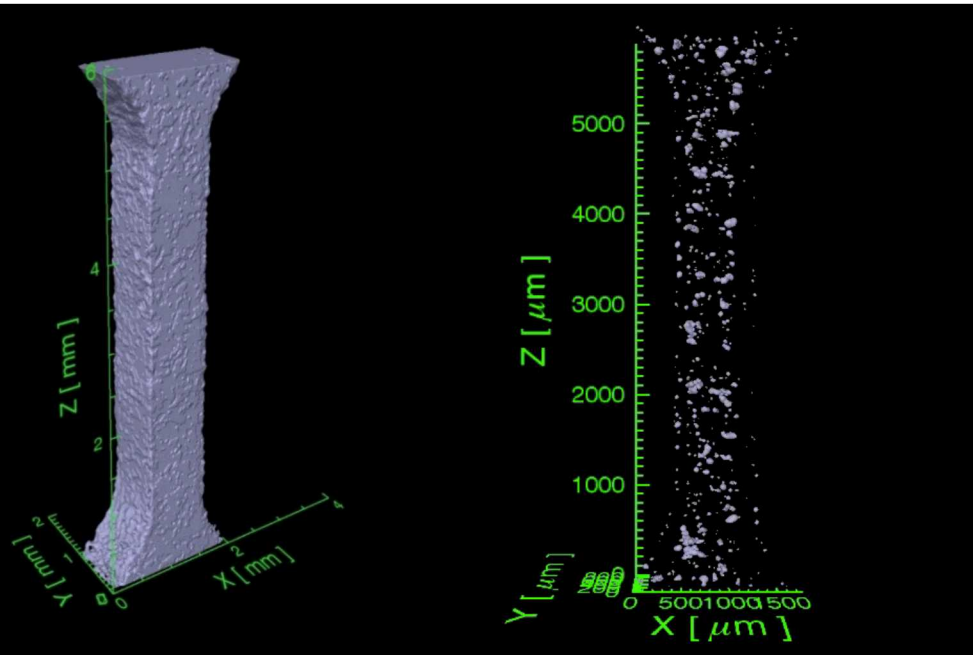


Build Scale Microstructure
T. Rodgers, J. Madison



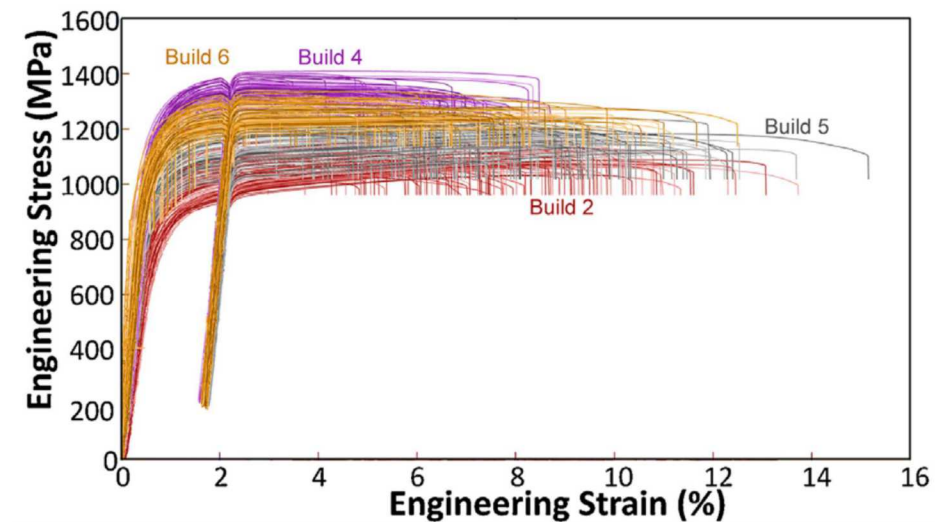
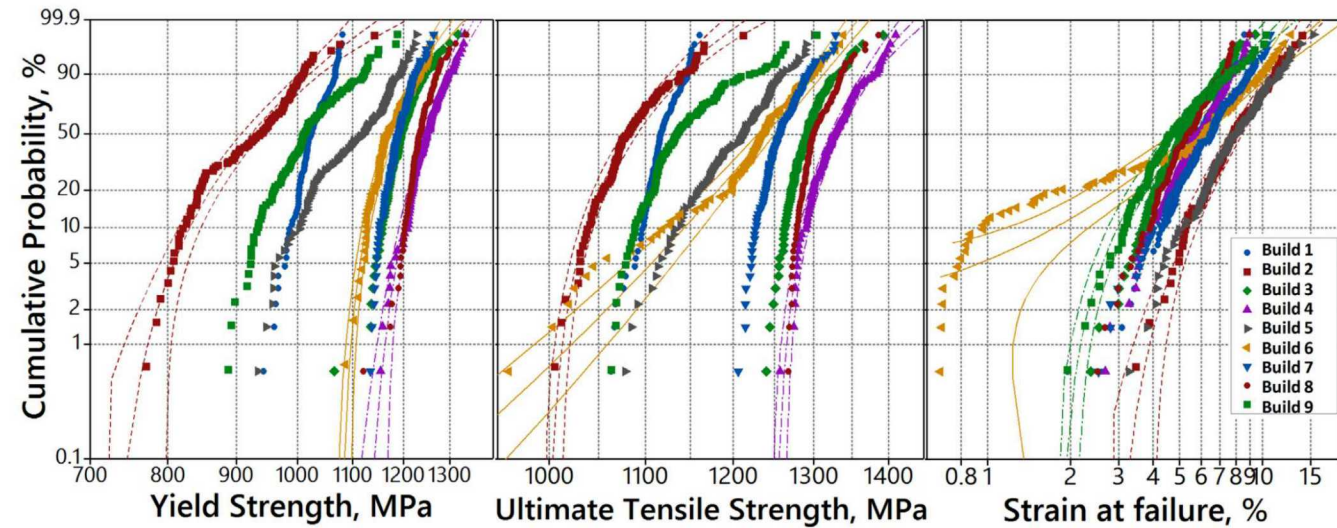
1. Motivation
2. Predicting AM Part Performance in the Sandia Fracture Challenge
3. Ongoing Work
4. Conclusions

Motivation: Additive Manufacturing (AM) Defects Can Lead to Large Performance Variability



(J. Madison, T. Ivanoff, D. Underwood, SNL)

Can we predict performance and failure of specific AM parts without explicitly meshing pores?



Sandia Fracture Challenge: Blind Predictions of Ductile Failure

Int J Fract (2014) 186:5–68
DOI 10.1007/s10704-013-9904-6

ORIGINAL PAPER

The Sandia Fracture Challenge: blind round robin predictions of ductile tearing

B. L. Boyce · S. L. B. Kramer · H. E. Fang · T. E. Cordova · M. K. Neilsen · K. Dion · A. K. Kaczmarowski · E. Karasz · L. Xue · A. J. Gross · A. Ghahremaninezhad · K. Ravi-Chandar · S.-P. Lin · S.-W. Chi · J. S. Chen · E. Yreux · M. Rüter · D. Qian · Z. Zhou · S. Bhamare · D. T. O'Connor · S. Tang · K. I. Elkhodary · J. Zhao · J. D. Hochhalter · A. R. Cerrone · A. R. Ingraffea · P. A. Wawrzynek · B. J. Carter · J. M. Emery · M. G. Veilleux · P. Yang · Y. Gan · X. Zhang · Z. Chen · E. Madenci · B. Kilic · T. Zhang · E. Fang · P. Liu · J. Lua · K. Nahshon · M. Miraglia · J. Cruce · R. DeFrese · E. T. Moyer · S. Brinckmann · L. Quinkert · K. Pack · M. Luo · T. Wierzbicki

Int J Fract (2016) 198:5–100
DOI 10.1007/s10704-016-0089-7

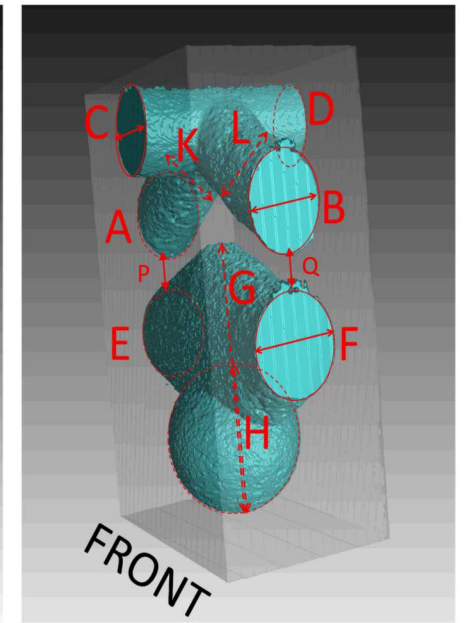
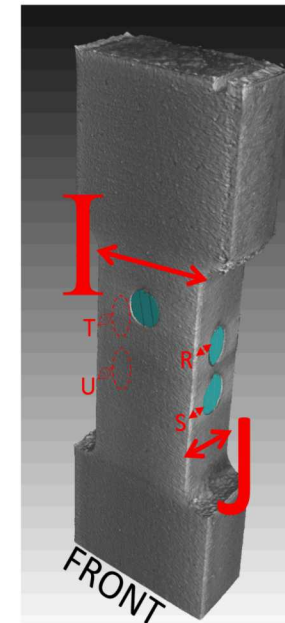
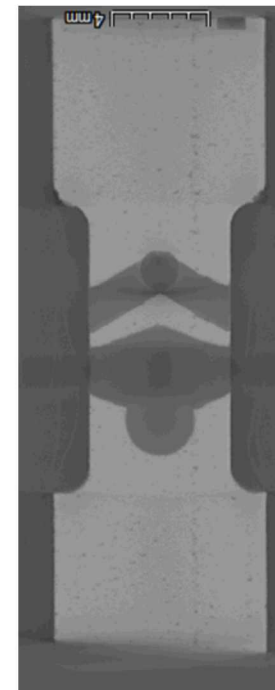
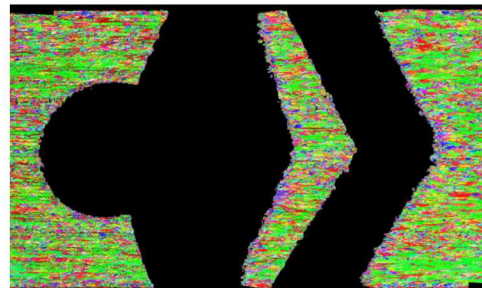
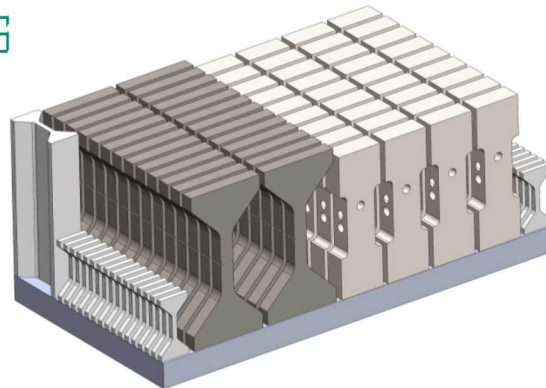
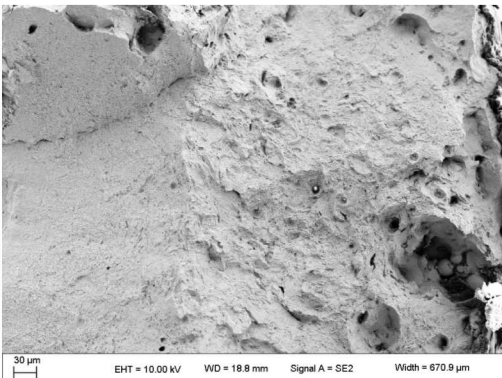
SANDIA FRACTURE CHALLENGE 2014

The second Sandia Fracture Challenge: predictions of ductile failure under quasi-static and moderate-rate dynamic loading

B. L. Boyce · S. L. B. Kramer · T. R. Bosiljevac · E. Corona · J. A. Moore · K. Elkhodary · C. H. M. Simha · B. W. Williams · A. R. Cerrone · A. Nonn · J. D. Hochhalter · G. F. Bomarito · J. E. Warner · B. J. Carter · D. H. Warner · A. R. Ingraffea · T. Zhang · X. Fang · J. Lua · V. Chiaruttini · M. Mazière · S. Feld-Payet · V. A. Yastrebov · J. Besson · J.-L. Chaboche · J. Lian · Y. Di · B. Wu · D. Novokshanov · N. Vajragupta · P. Kucharczyk · V. Brinell · B. Döbereiner · S. Münstermann · M. K. Neilsen · K. Dion · K. N. Karlson · J. W. Foulk III · A. A. Brown · M. G. Veilleux · J. L. Bignell · S. E. Sanborn · C. A. Jones · P. D. Mattie · K. Pack · T. Wierzbicki · S.-W. Chi · S.-P. Lin · A. Mahdavi · J. Predan · J. Zdravcevic · A. J. Gross · K. Ravi-Chandar · L. Xue

The Third Sandia Fracture Challenge – 2016

- 316L AM Part
- Goal: Predict Tensile Failure
- 21 Participant Teams



Bammann-Chiesa-Johnson (BCJ) Constitutive Model for Plasticity

- Based on work by Bammann et al. 1993, Brown and Bammann 2012
- History-dependent viscoplastic internal state variable model
- Stress is dependent on damage ϕ and evolves according to

$$\dot{\sigma}_{ij} = \left(\frac{\dot{E}}{E} - \frac{\dot{\phi}}{1 - \phi} \right) \sigma_{ij} + E(1 - \phi)(\dot{\epsilon}_{ij} - \dot{\epsilon}_{ij}^p)$$

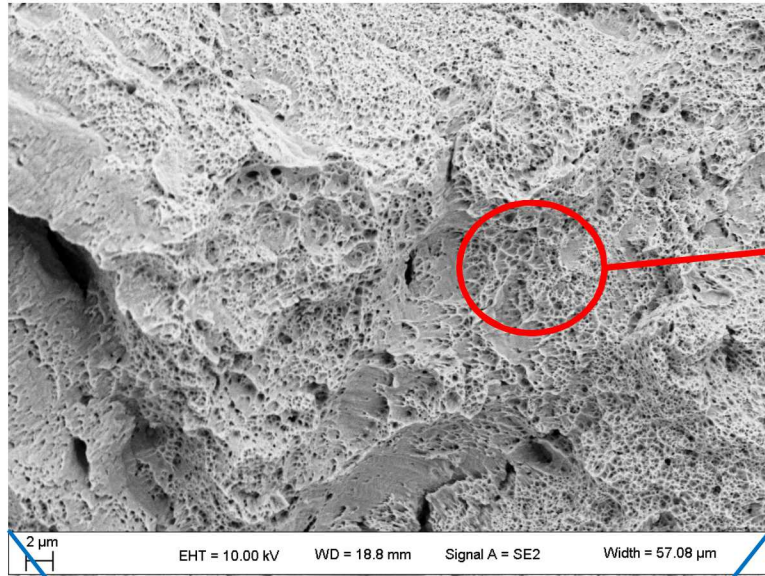
- Flow rule includes yield stress and internal state variable for hardening

$$\dot{\epsilon}_p = \sinh \left\langle \frac{\sigma_e}{Y + \kappa} - 1 \right\rangle$$

- The isotropic hardening variable κ evolves in a hardening minus recovery form.

$$\dot{\kappa} = \kappa \frac{\dot{\mu}}{\mu} + (H - R_d \kappa) \dot{\epsilon}_p$$

Fracture Surfaces Indicated Both Existing Pores and Pore Nucleation



Void Nucleation

Fine scale voids ($< 1\mu\text{m}$) indicate nucleation

$$\dot{\eta} = \eta \dot{\epsilon}_p \left(N_1 \left[\frac{4}{27} - \frac{J_3^2}{J_2^3} \right] + N_2 \frac{J_3}{J_2^3} + N_3 \frac{\langle p \rangle}{\sigma_e} \right)$$

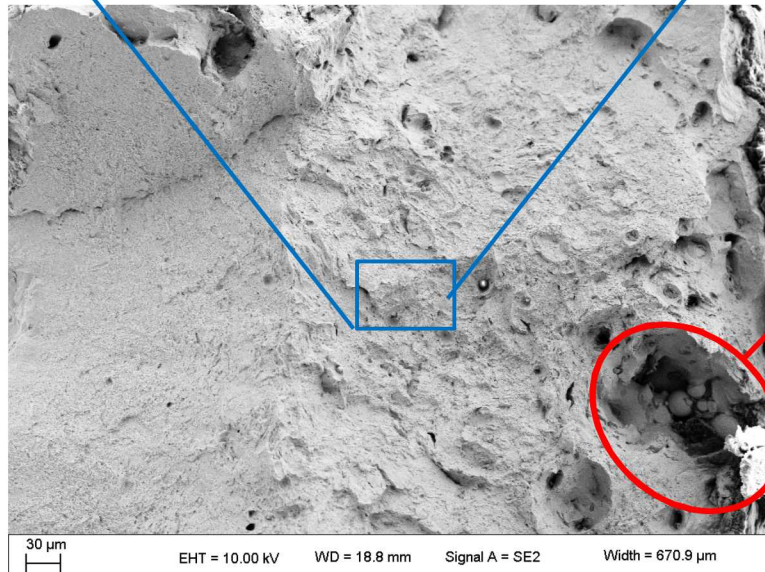
Void Growth

Pre-existing voids captured by void growth

$$\dot{\phi} = \sqrt{\frac{2}{3}} \dot{\epsilon}_p \frac{1 - (1 - \phi)^{m+1}}{(1 - \phi)^m} \sinh \left[\frac{2(2m - 1)}{2m + 1} \frac{\langle p \rangle}{\sigma_e} \right]$$

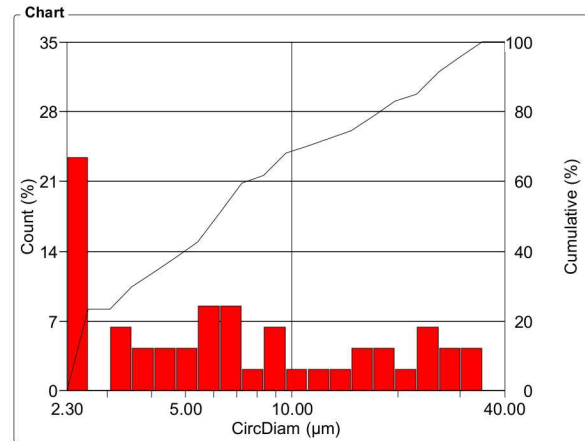
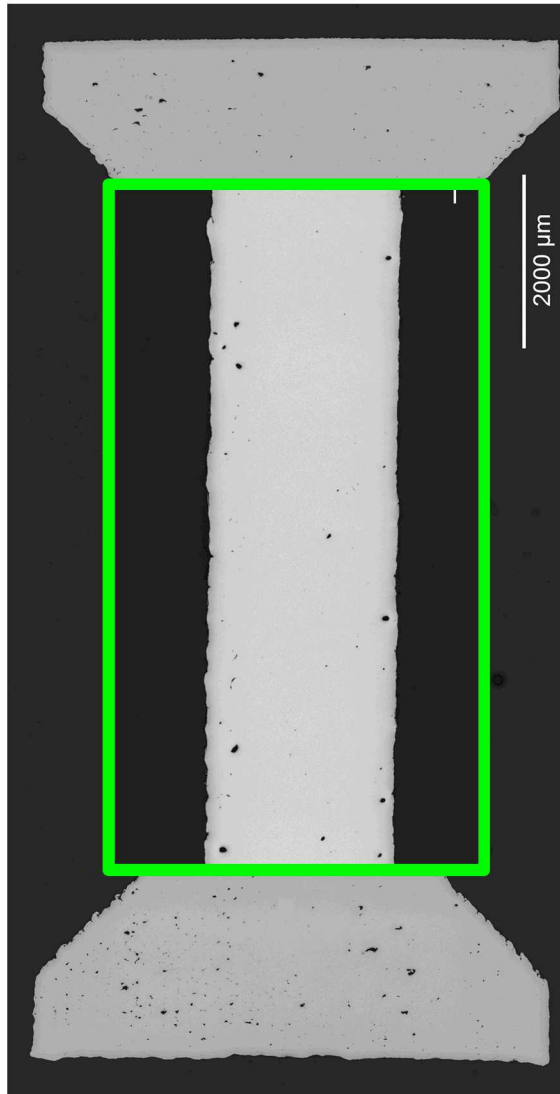
Total Damage

$$\phi = \frac{\eta v_v}{1 + \eta v_v}$$



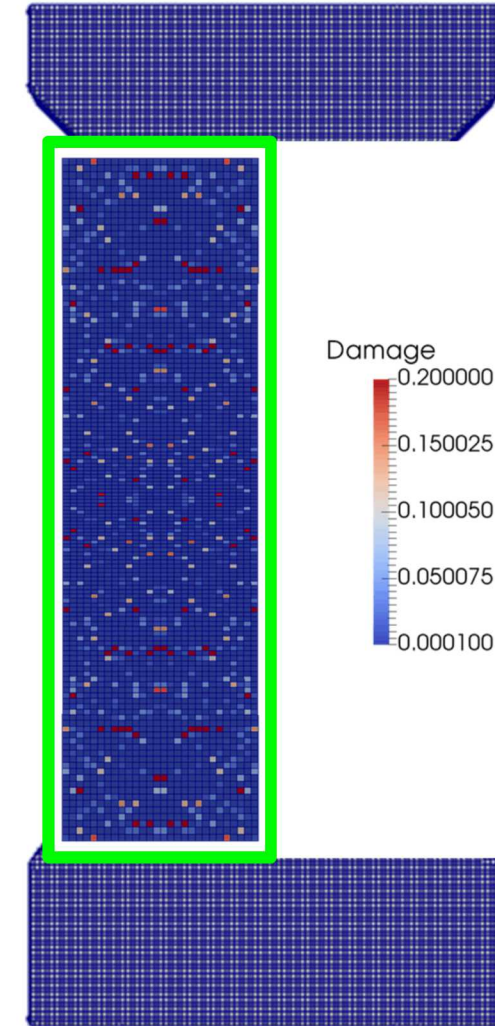
(Horstemeyer & Gokhale 1999)

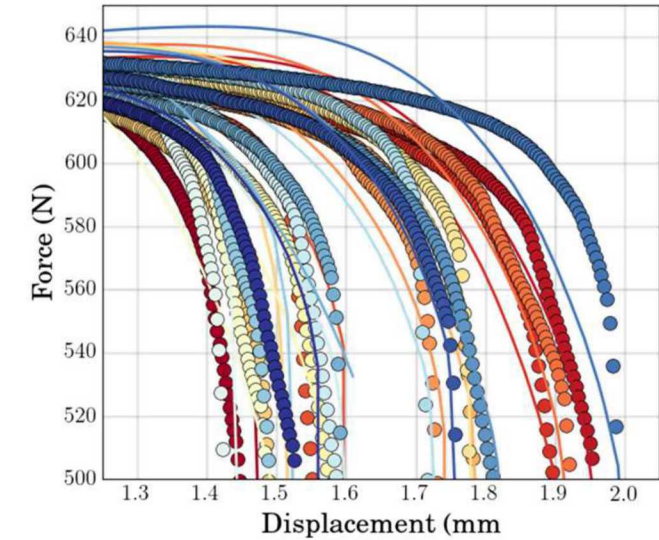
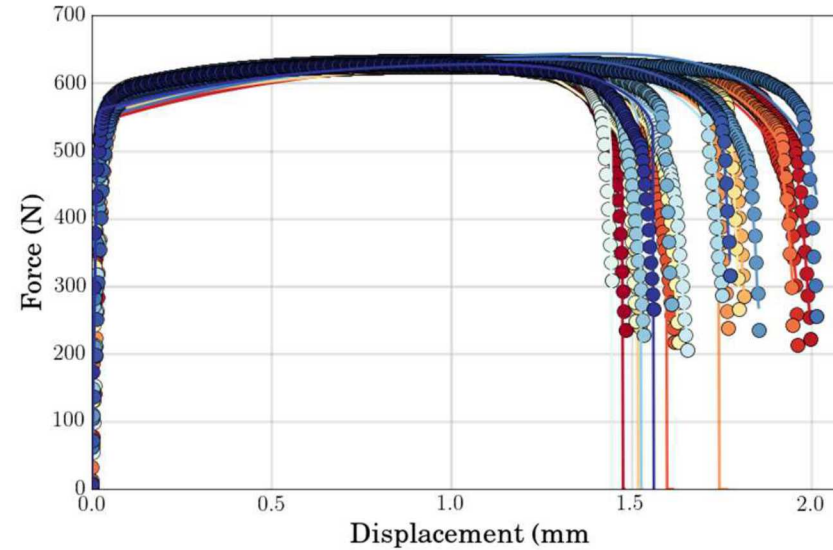
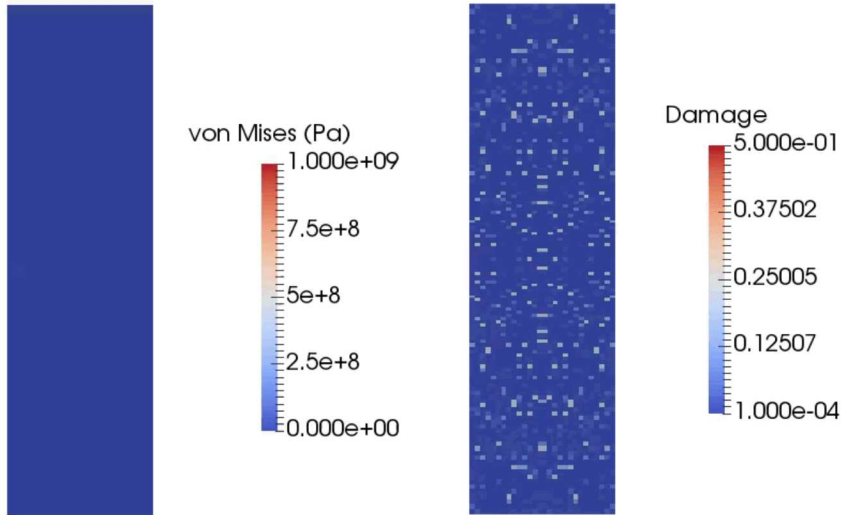
8 Incorporating Porosity as Initial Damage



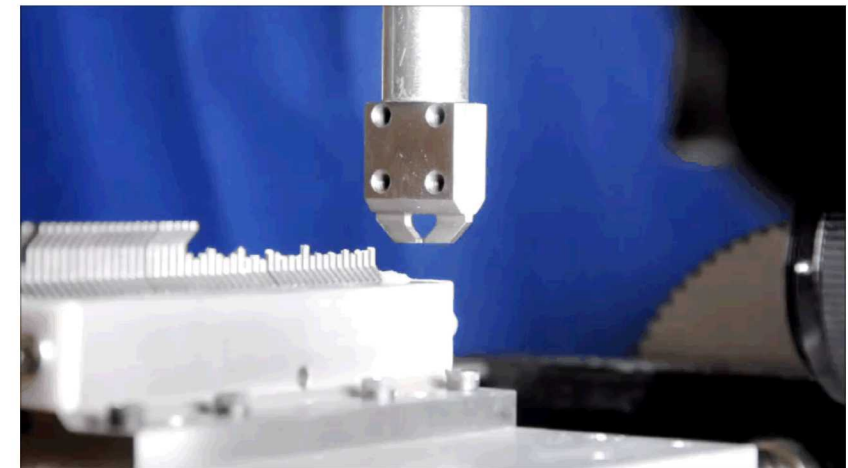
Porosity Mapping

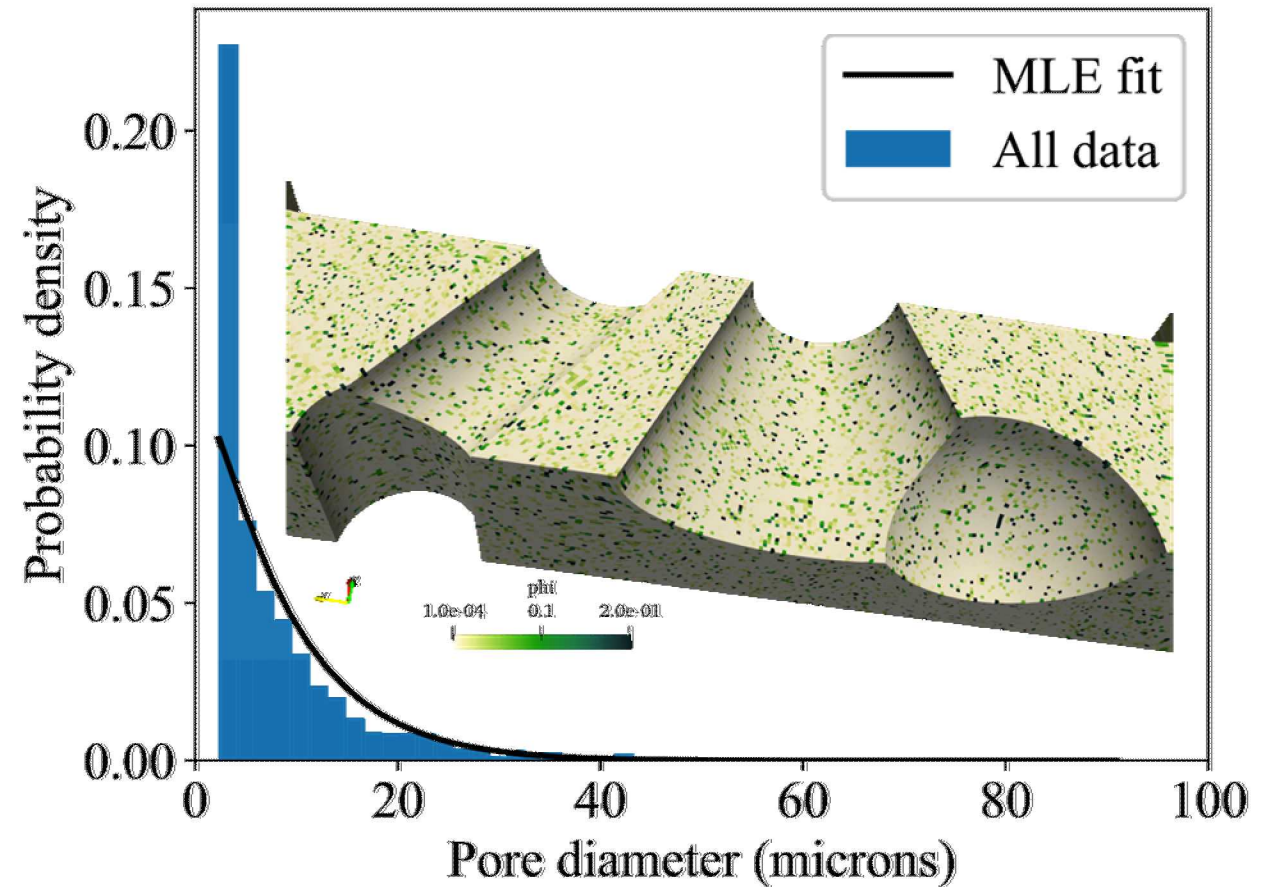
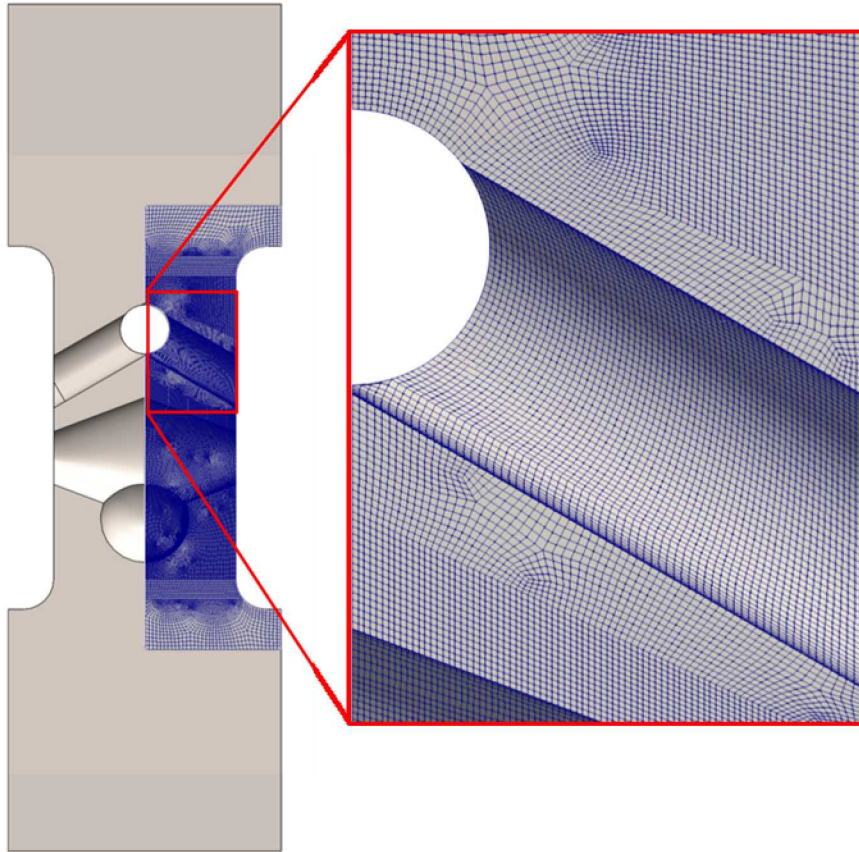
x, y, z, r_{pore}



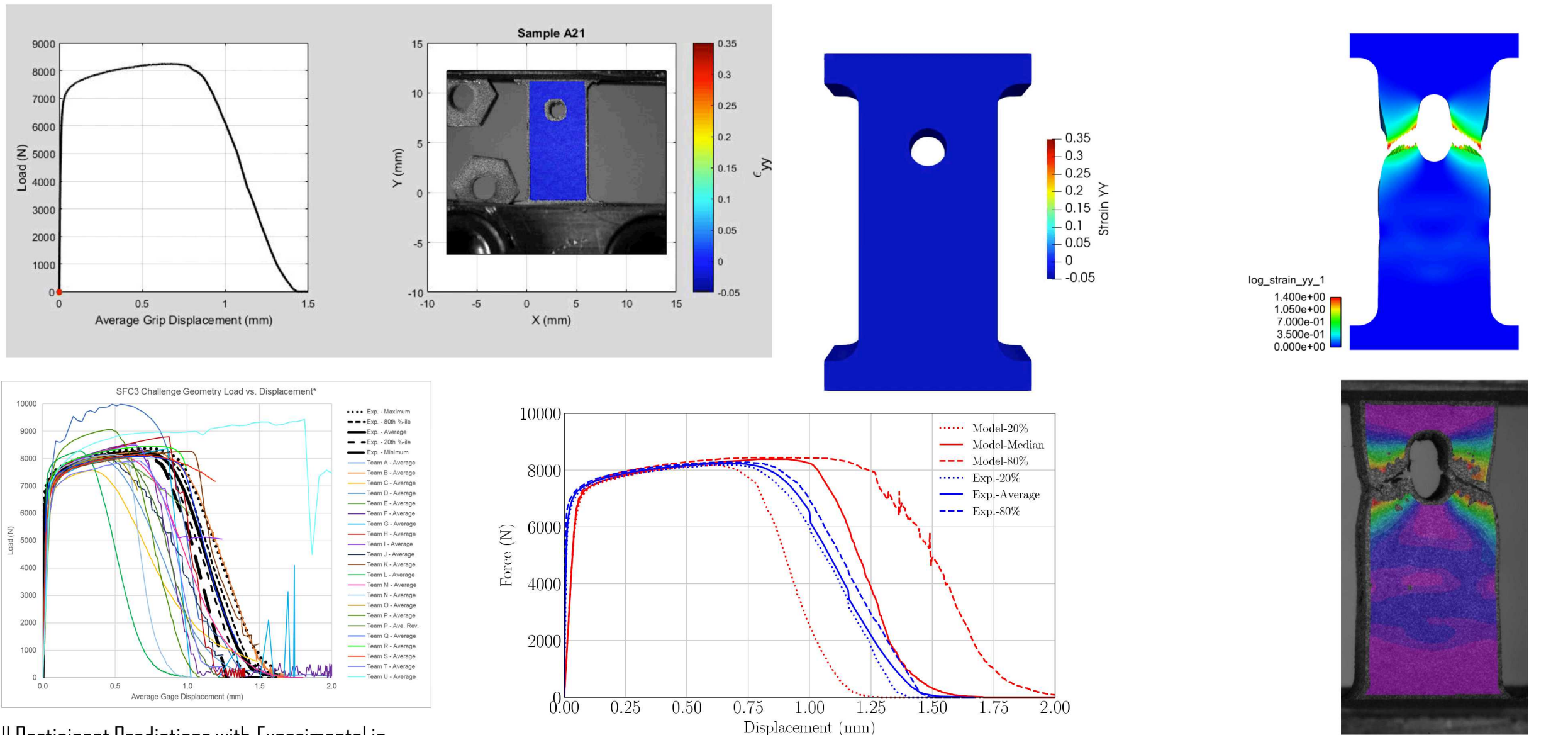


- A parameter set was calibrated for each longitudinal tension test
- Transverse and notched tensile data were not used due to time constraints
- FEA performed in Sierra/SM
- Calibration performed using Dakota and MatCal (Kyle Karlson)





- Multiple porosity initializations were used to capture uncertainty in performance

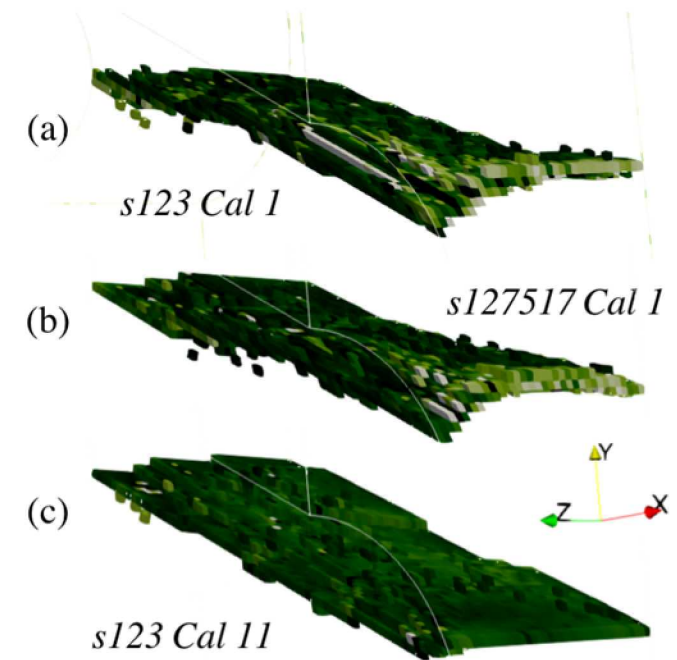
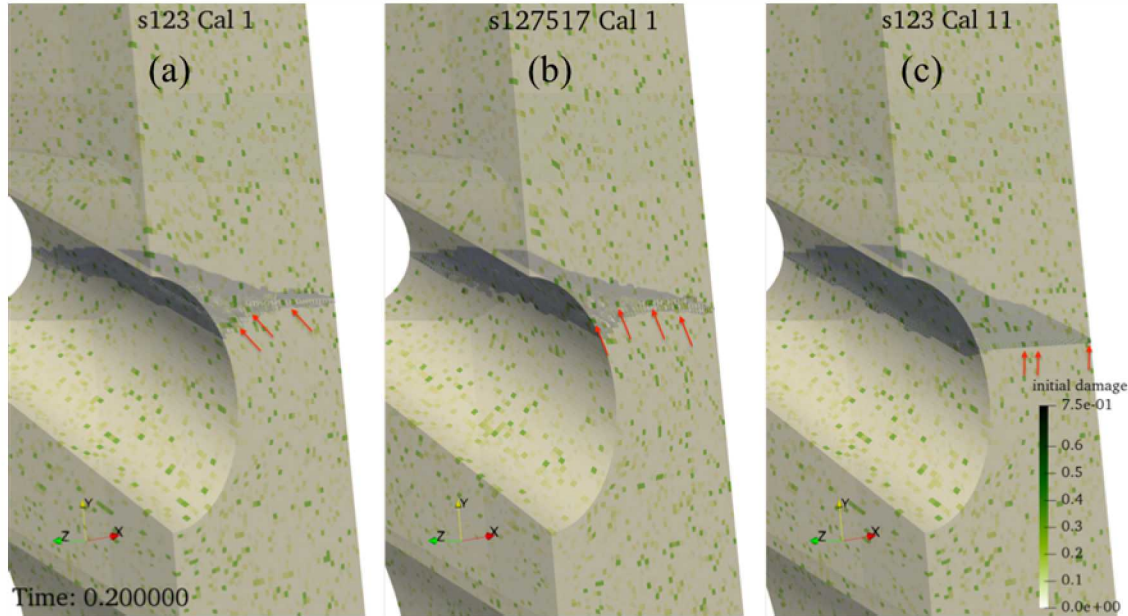


All Participant Predictions with Experimental in Black

SNL/NM Prediction

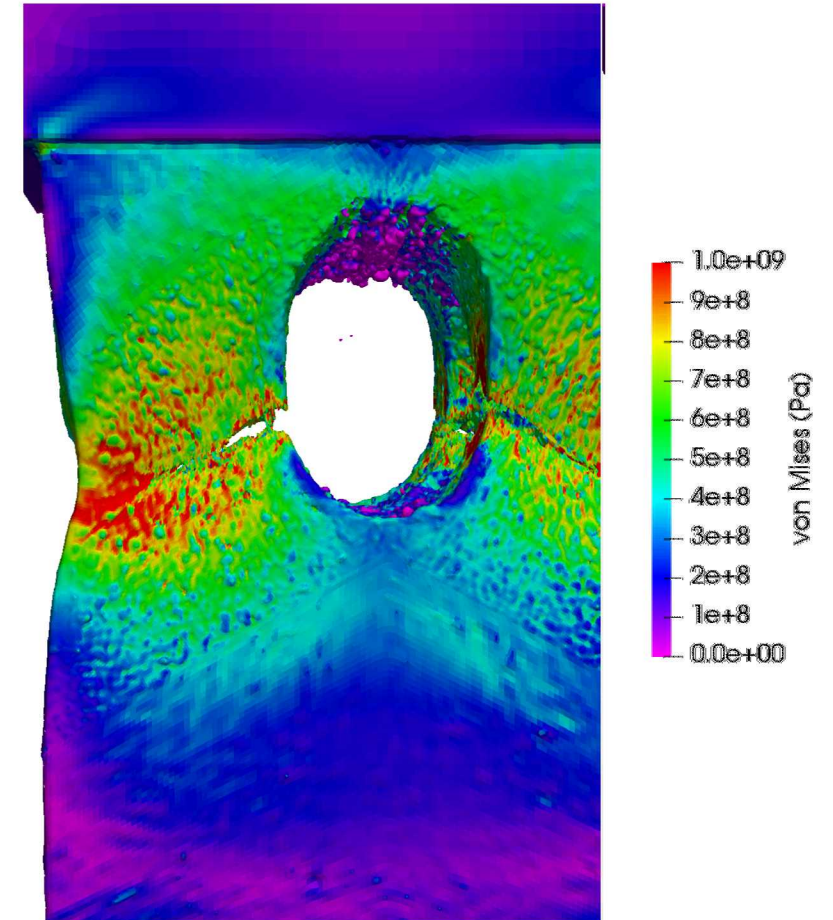
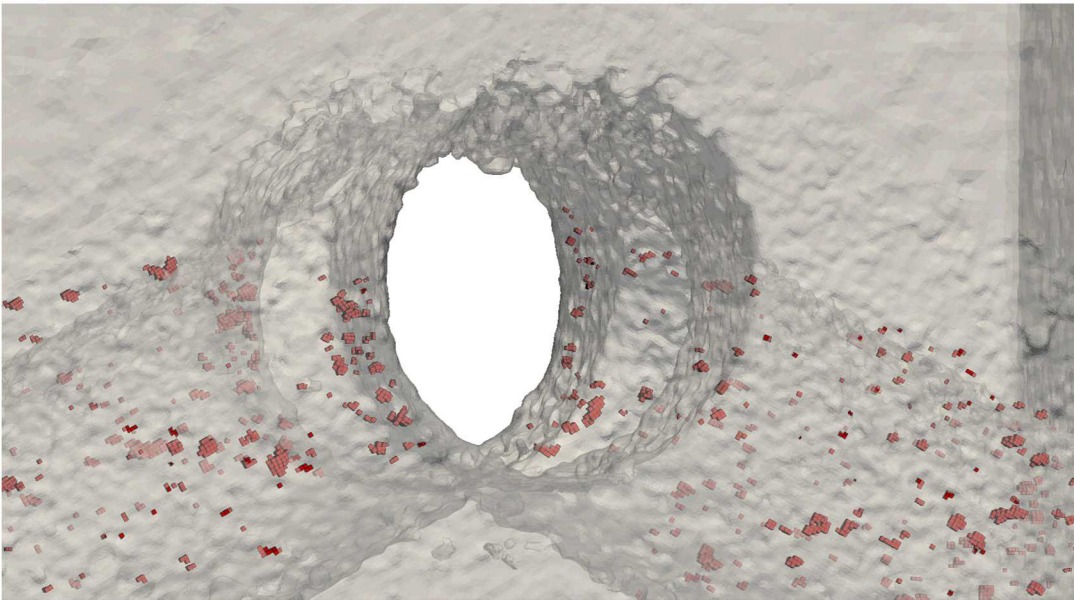
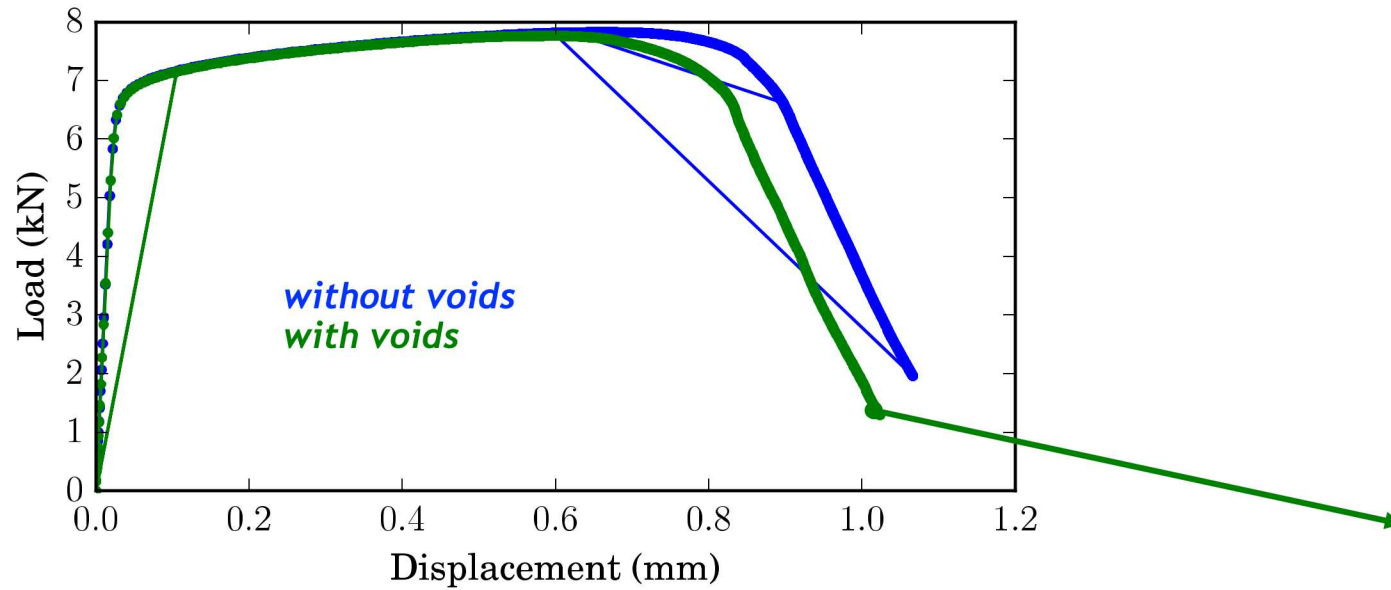
Johnson et al. IJF (Accepted), Kramer, Boyce et al., IJF (Accepted)

Different Porosity Distributions Affect Crack Path



- Porosity seed indicated by s number, i.e. $s123$ is a different realization than $s127517$
- For the same calibration number (Cal 1), a different porosity seed yields a different crack path

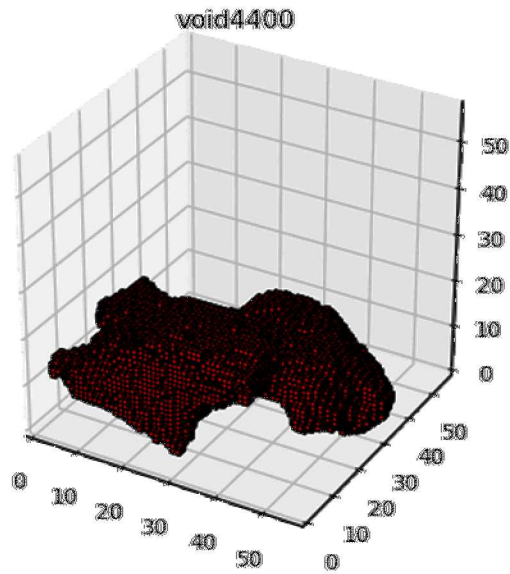
Modeling the Effect of Pores and Surface Roughness From High-Resolution CT Scan in Follow-up Investigation



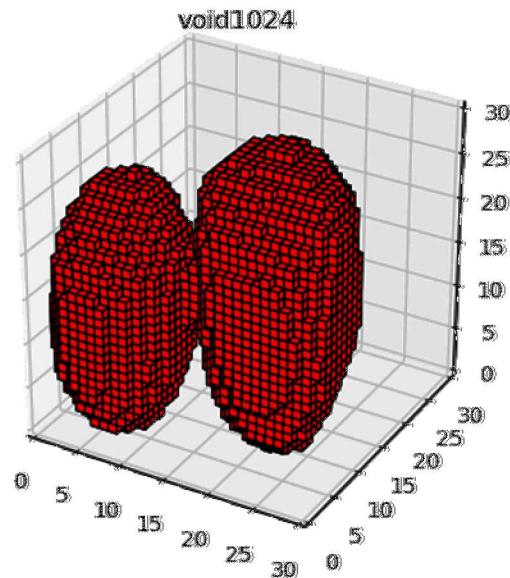
Simulation results from ~13 million element meshes generated from CT scans.

RVE Simulations Investigating Pore Effects

A Tale of Two CT's

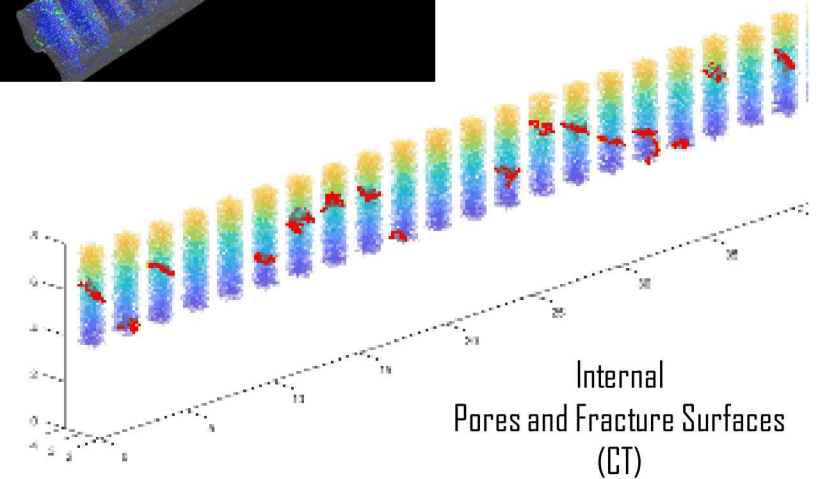
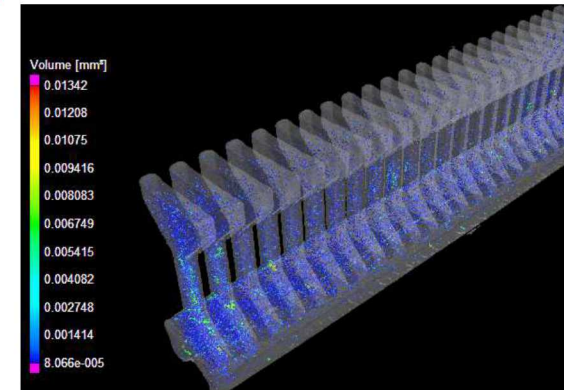


29 um resolution
1.9 um voxel size

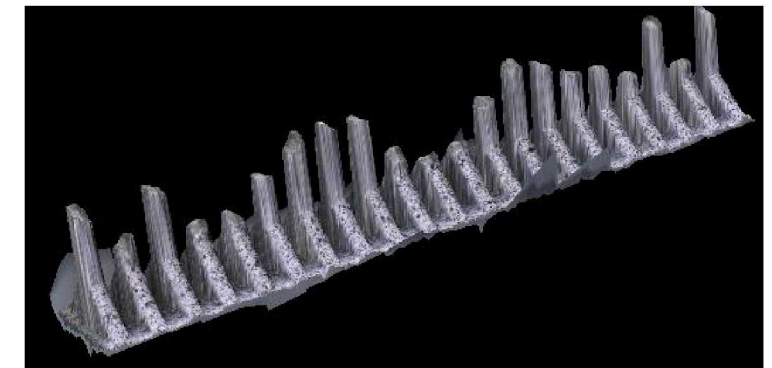


15.6 um resolution
2.5 um voxel size

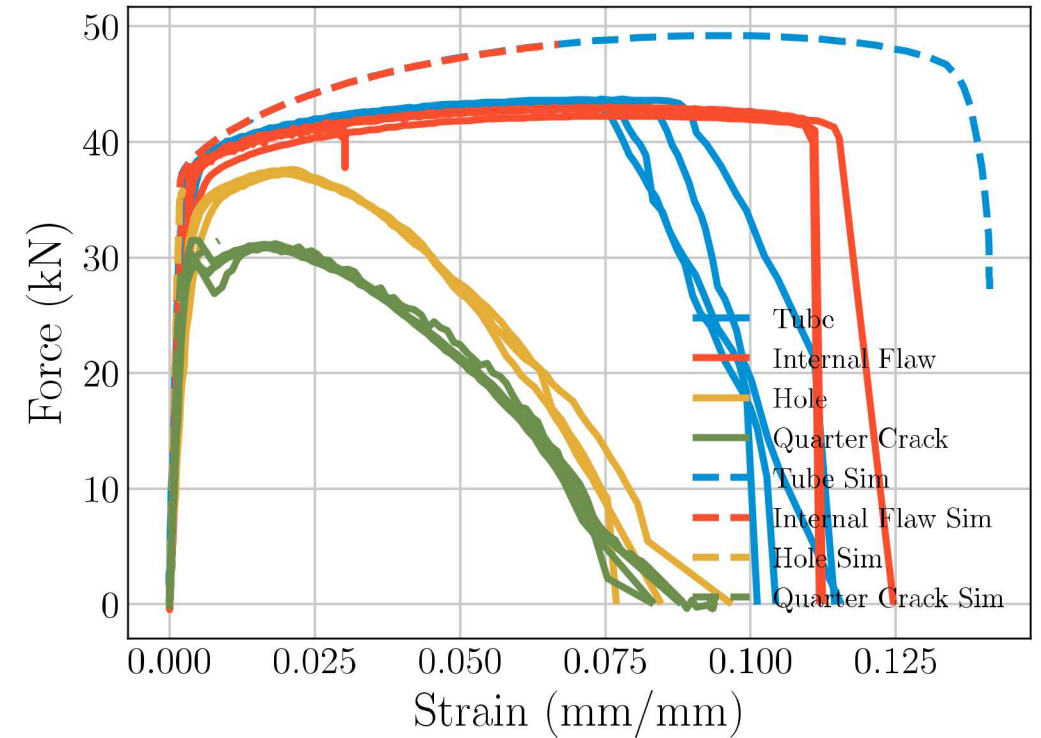
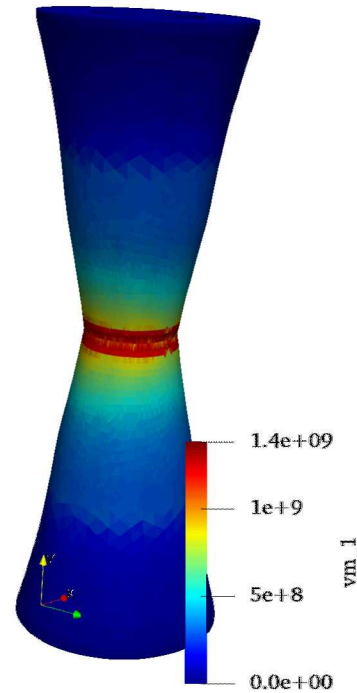
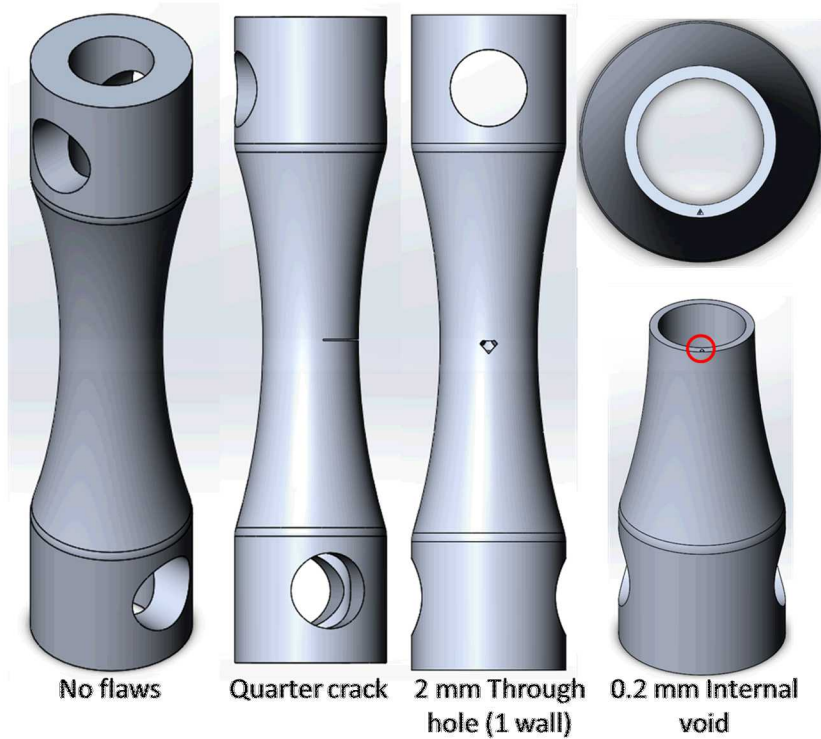
Reproducing Tension Specimen Failure



Internal
Pores and Fracture Surfaces
(CT)



- Approach: Refine mesh of parts from high resolution CT scans to approach limit of explicitly meshed pores (many elements within a pore)
- Where does populating high local damage for pore locations break down?



- Hollow LPBF 316L stainless steel tubes have different “defects” designed into wall thickness:
 - 1/4 diameter crack
 - 2mm cube through-hole
 - 0.2 mm cubic internal void
- Internal void design has greater ductility than tube with no flaw > design can control performance

- AM Materials often have significant material variability
- Part performance and failure can be predicted using existing tools
 - Part geometry can be used as a tool for flaw tolerance
- Significant scatter in tension data may not translate to scatter in part performance
- Porosity can possibly be accounted for using a damage formulation
- Different defect structures affect crack initiation and propagation

Future Directions

- Further examining using local damage model to represent spatial porosity
- Investigating effects of surface roughness
- What can be done with coarse CT scans?

