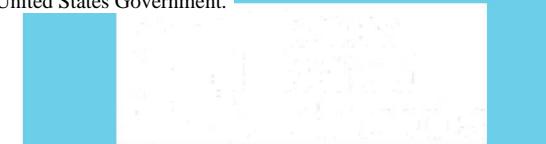
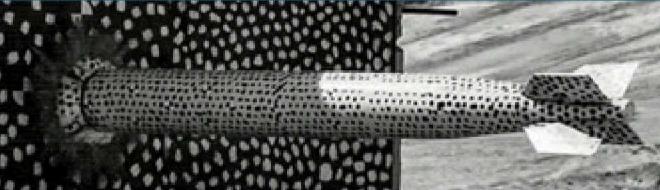


# Infrared absorption oscillator strength factors in $\text{SiN}_x$ thin films



*PRESENTED BY*

Sara DiGregorio

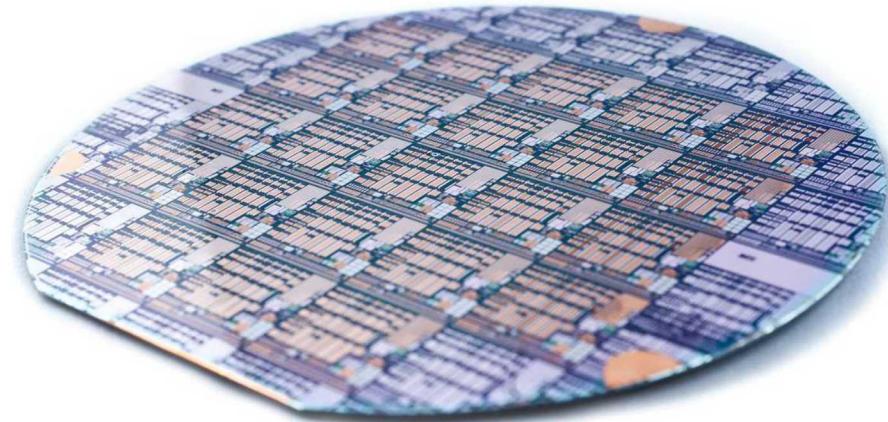


Sandia National Laboratories is a multimission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

# Background

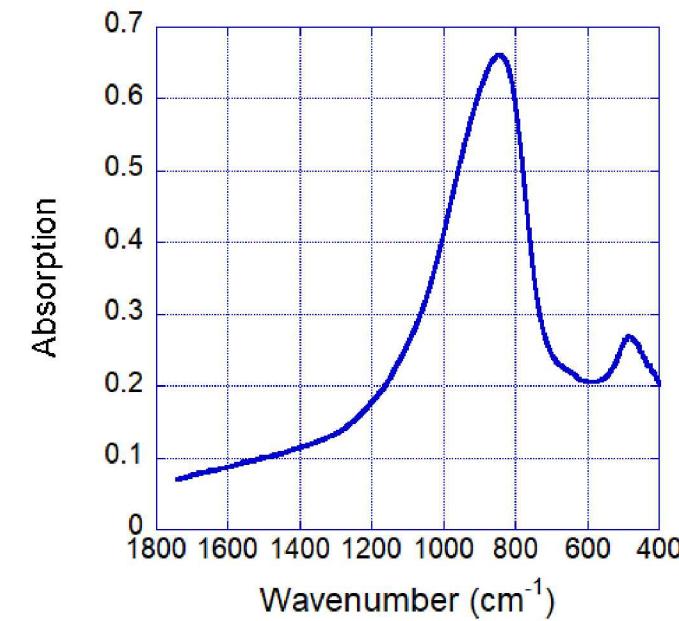
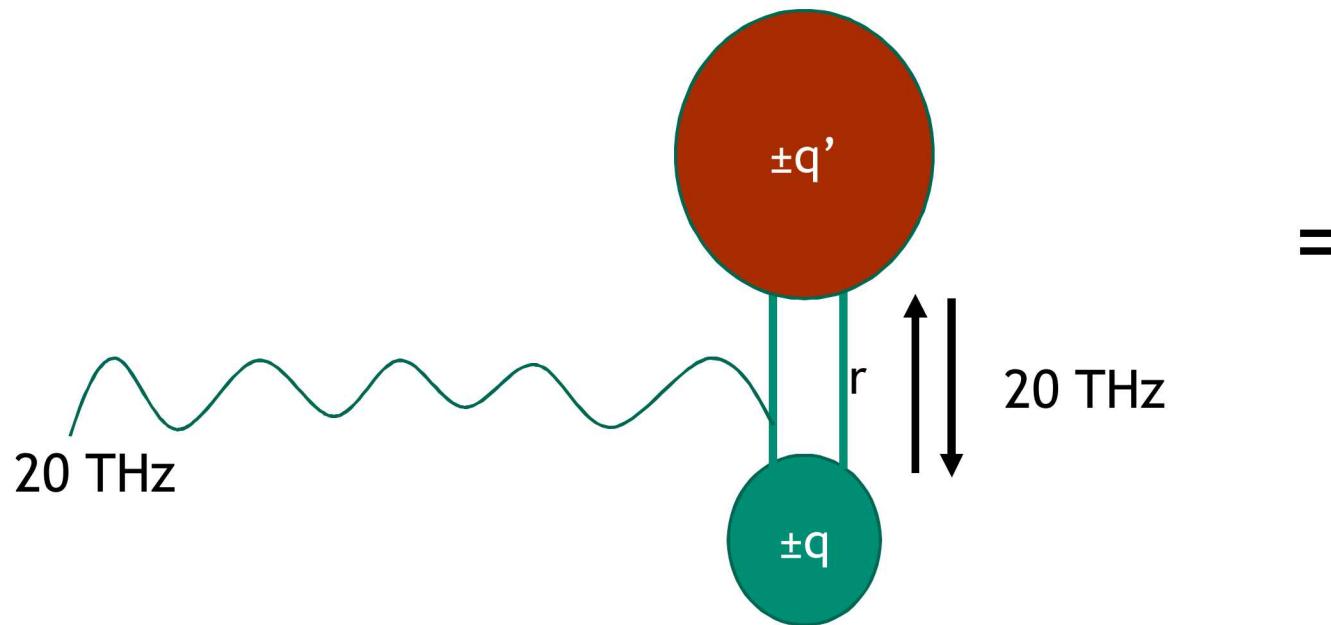
## Silicon nitride, $\text{SiN}_x$ , is used in microelectronics fabrication

- Chemically and thermally stable
- Useful dielectric properties
- Many deposition techniques available
- Many properties have been published
- The oscillator strength factor (K) of the  $\text{SiN}_x$  asymmetric stretch mode is the subject of this study
- We used IR absorption to characterize the K value for each film



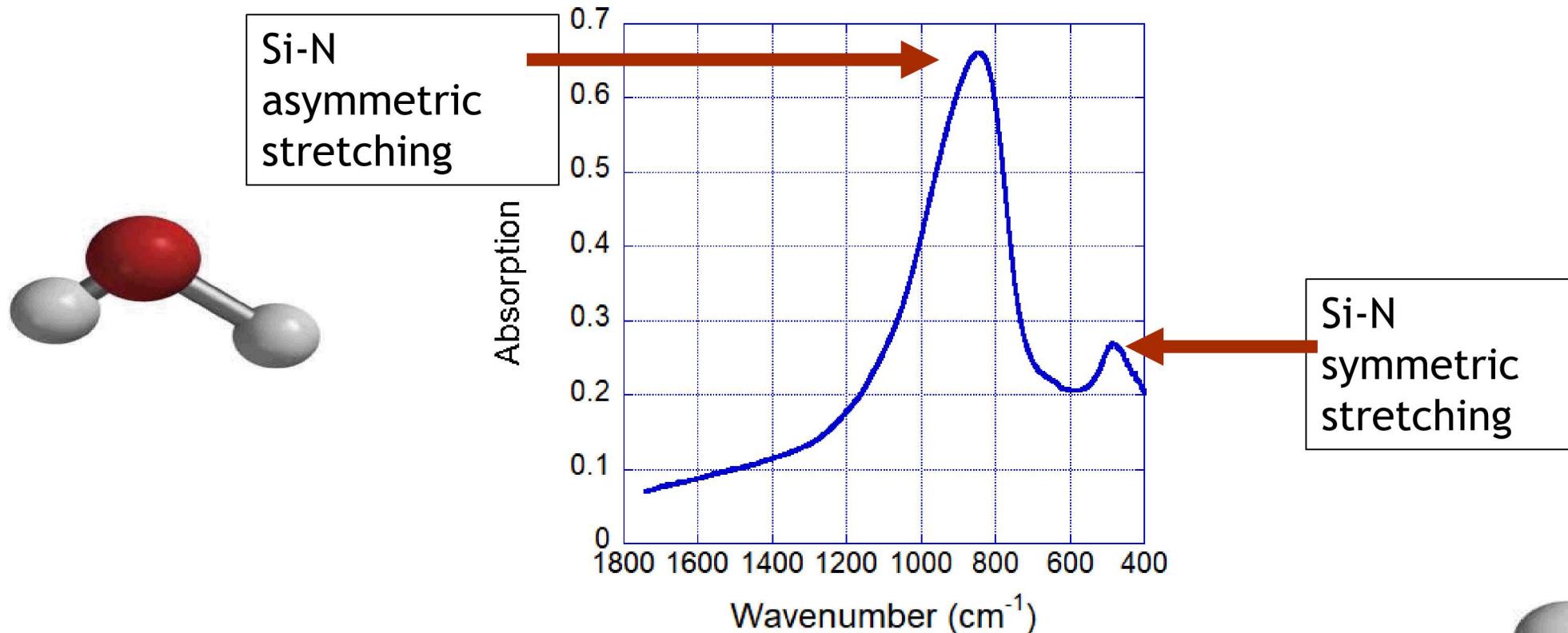
## Absorption occurs when an oscillating dipole interacts with light of the same frequency

- Bonds vibrate at natural resonating frequencies
- Bond dipole =  $q^*r$
- If the incoming light wave oscillates at the same frequency as the molecular dipole, absorption occurs

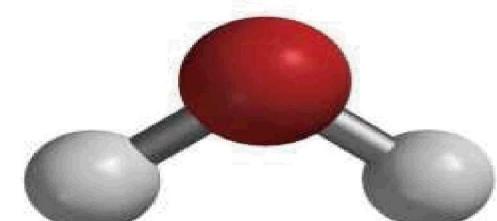


## The oscillator strength factor, K is characterized by Si-N infrared active asymmetric stretch mode

- The two IR active Si-N modes are symmetric stretching centered about  $480\text{ cm}^{-1}$  and asymmetric stretching centered about  $830\text{-}875\text{ cm}^{-1}$

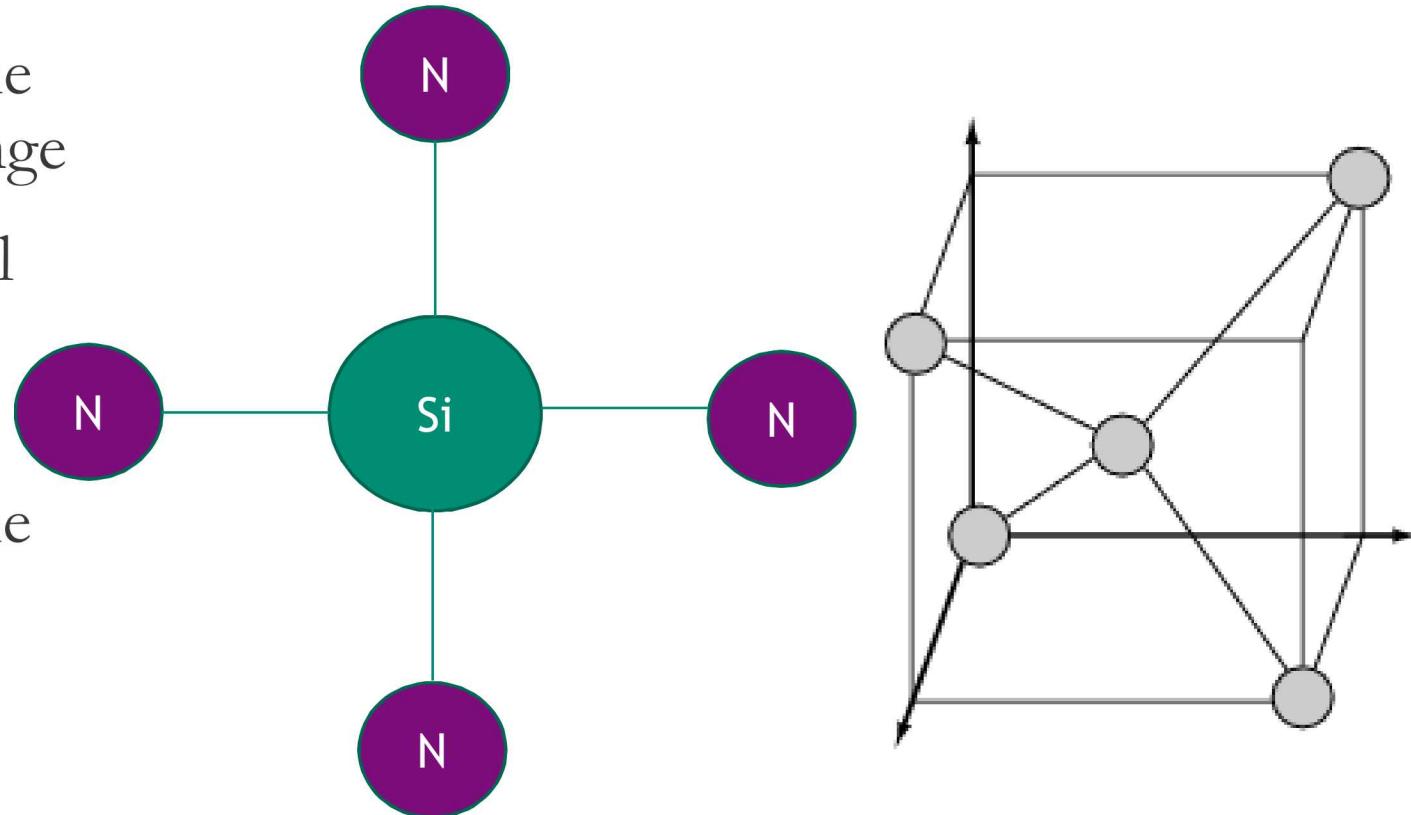


Absorption spectra of a stoichiometric  $\text{SiN}_x$  film of 142.7 nm



## 6 The Si-N asymmetrical stretch mode is sensitive to changes in Si content

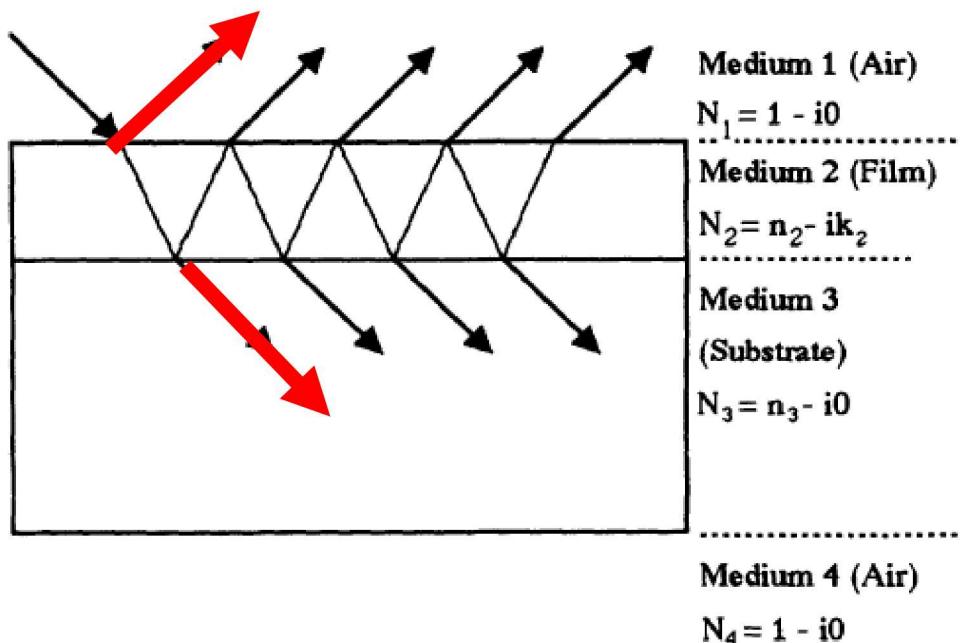
- An increase in Si content causes the  $\text{SiN}_x$  tetrahedron structure to change
- Compositionally induced structural changes alter the IR properties
- We wanted to see how the IR properties changed with altering the local bonding structure by varying film composition and thickness



Stoichiometric  $\text{Si}_3\text{N}_4$   
structural tetrahedron

# Multiple reflections must be accounted for in thin film spectroscopy

- IR measurements are used to characterize the compositionally induced changes to the Si-N asymmetric stretch mode
- Multiple reflections occur at the film substrate interface and within the substrate



$$T = \frac{(1 - R_{12})(1 - R_{23})(1 - R_{34})e^{-ad}}{(1 - R_{23}R_{34}) - (R_{12}R_{23} + R_{12}R_{34} - 2R_{12}R_{23}R_{34})e^{-2ad}}.$$

Full expression for transmittance  
with multiply reflected beams

Multiply reflected beams off the  
film substrate interface

Maley, N. (1992). Physical  
Review B, 46(4) 2078-2085

## Absorption coefficient must be defined in terms of transmission and reflection to account for multiple reflections

- The absorption coefficient is solve from IR transmission where reflections at the film substrate interface are negligible and reflections within the substrate are incoherent

$$T(\omega) = \frac{(1 - R(\omega))^2 e^{-\alpha(\omega)d}}{1 - R(\omega)^2 e^{-2\alpha(\omega)d}}$$

- Thus,  $\alpha(\omega)$  is then given by

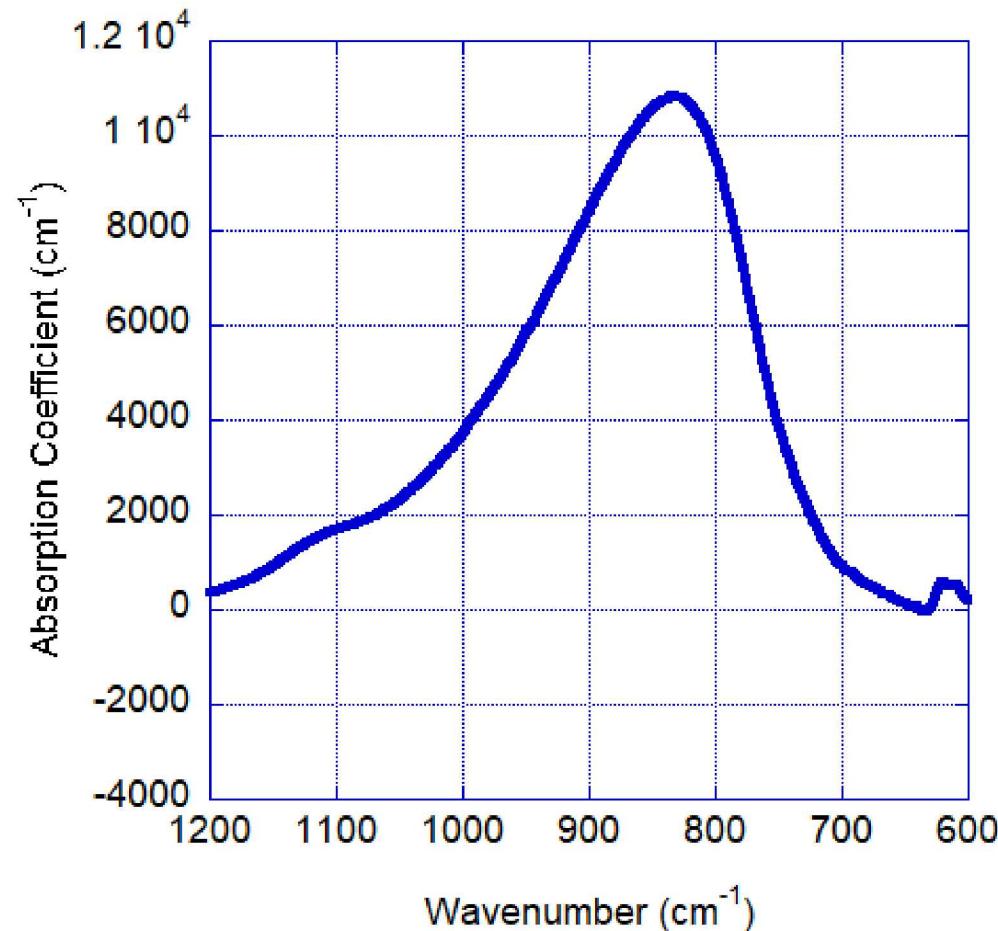
$$\alpha(\omega) = \frac{-1}{d} \ln\left(\frac{A(\omega)^2}{T(\omega)} \left[ \left(1 + \left(\frac{T(\omega)}{A(\omega)^2 R(\omega)}\right)^2\right)^{1/2} - 1 \right]\right)$$

- where

$$A(\omega)^2 = \frac{(1 - R(\omega))^2}{2R(\omega)^2}$$

Brodky, M.H., Cardona, M. & Cuomo, J.J. (1977). Physical Review B, 16(8), 3556-3571. Stavola, M. & Fowler, W.B. (2018). Journal of Applied Physics, 123.

## Each film has a characteristic absorption coefficient



Absorption coefficient for a  $\text{Si}_3\text{N}_4$  film  
238.2 nm thick

- Each film has a characteristic absorption coefficient that takes into account multiple reflections
- The integrated absorption coefficient is what is used to solve for K

The oscillator strength factor (K) is a parameter that relates bond density to the integrated absorption coefficient

$$[N] = \frac{c^2 n \omega_t \mu}{\pi q^2} \int \frac{a(\omega)}{\omega} d\omega$$

- $[N]$ =nitrogen bond concentration
- $n$ =refractive index
- $\omega_t$ =peak frequency of the Si-N asymmetric stretch mode
- $\mu$ =reduced mass
- $q$ =effective oscillating charge
- $\alpha(\omega)$ =absorption coefficient

$$K \equiv \frac{c^2 n \omega_t \mu}{\pi q^2}$$

$$K = \frac{[N]}{\int \frac{\alpha(\omega)}{\omega} d\omega}$$

Oscillator strength factor (K)

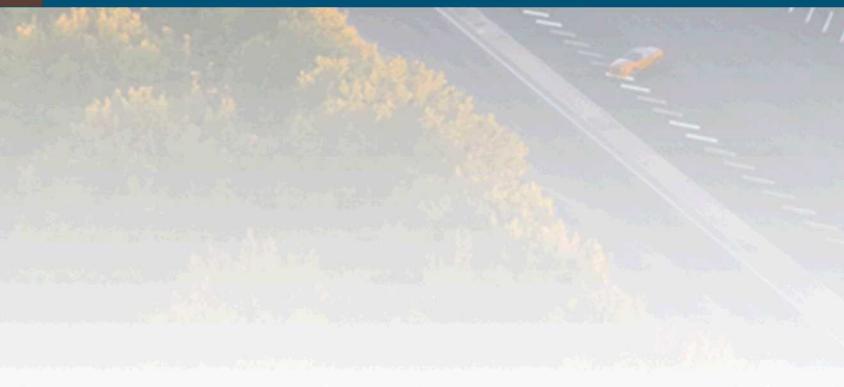
Motivation for characterizing K was to find a universal parameter that could be used to solve for [N] for any  $\text{SiN}_x$  film

$$K = \frac{[N]}{\int \frac{\alpha(\omega)}{\omega} d\omega}$$

- [N] could be found from film thickness and IR measurements alone if K was constant for every film composition



# Experimental

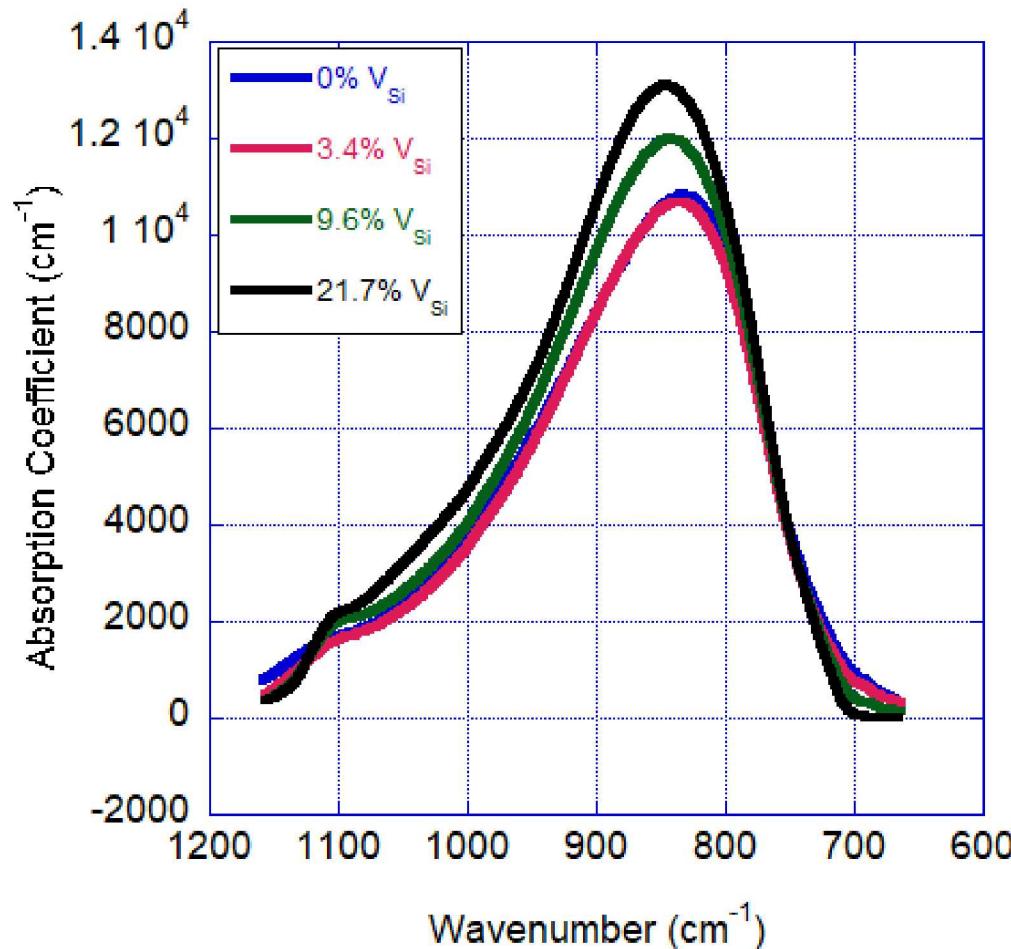


## $\text{SiN}_x$ LPCVD films were prepared in a vertical thermal reactor for films of varying composition and thickness

- Low pressure chemical vapor deposition (LPCVD)
- Dichlorosilane and ammonia
- $3\text{SiH}_2\text{Cl}_2 + \text{NH}_3 \rightarrow \text{Si}_3\text{N}_4 + 3\text{HCl} + 6\text{H}_2$
- Composition ranged from 0-25% volume fraction silicon ( $V_{\text{Si}}$ )
- Thickness ranged from 136-558 nm
- $[\text{N}]$  was calculated for each film from density and ellipsometry measurements
- T and R were collected for each film
- Absorption coefficient was calculated for each film

# Results

# Increasing $V_{Si}$ leads to less tensile film strain



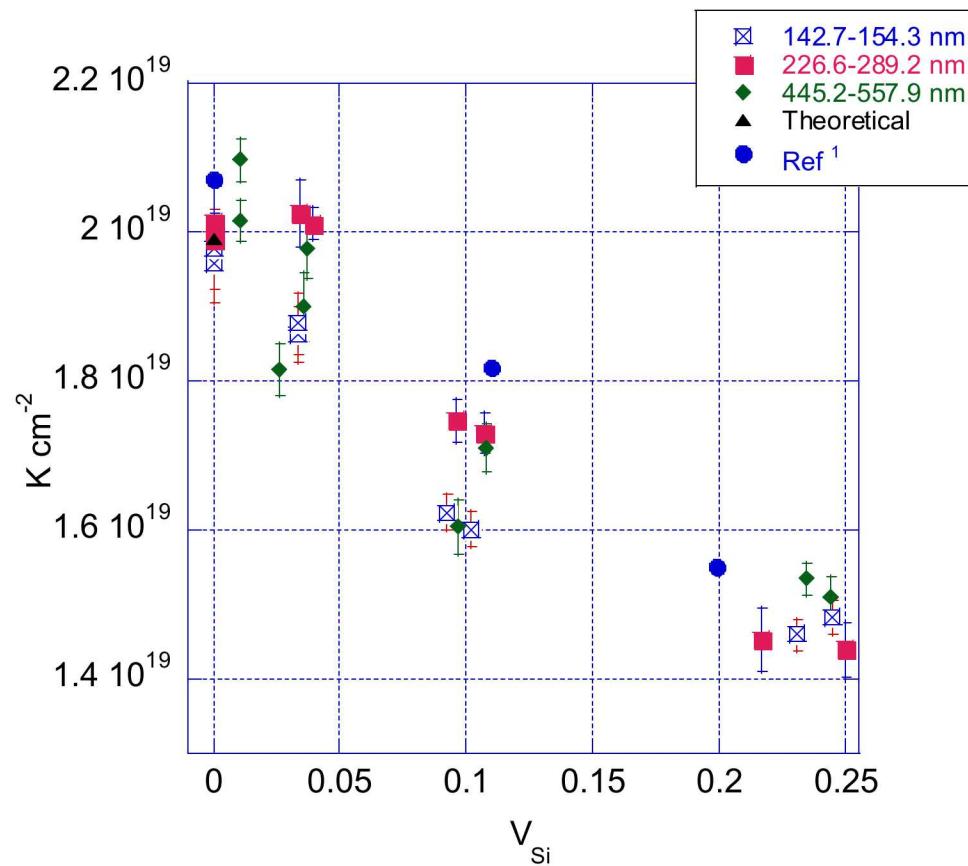
Absorption coefficient for SiNx films  
in the 226-289 nm thickness range  
and varying compositions

- Peak shifts to higher frequencies with increasing  $V_{Si}$
- $\varepsilon$  is the intrinsic bond strain
- $\nu_1$  is the wavenumber associated with a reference condition ( $848 \text{ cm}^{-1}$ )
- $\nu_2$  is film peak wavenumber

$$\varepsilon = \frac{\nu_1 - \nu_2}{\nu_1}$$

Central force model

# An increase of $V_{Si}$ leads to a decrease in oscillator strength factor

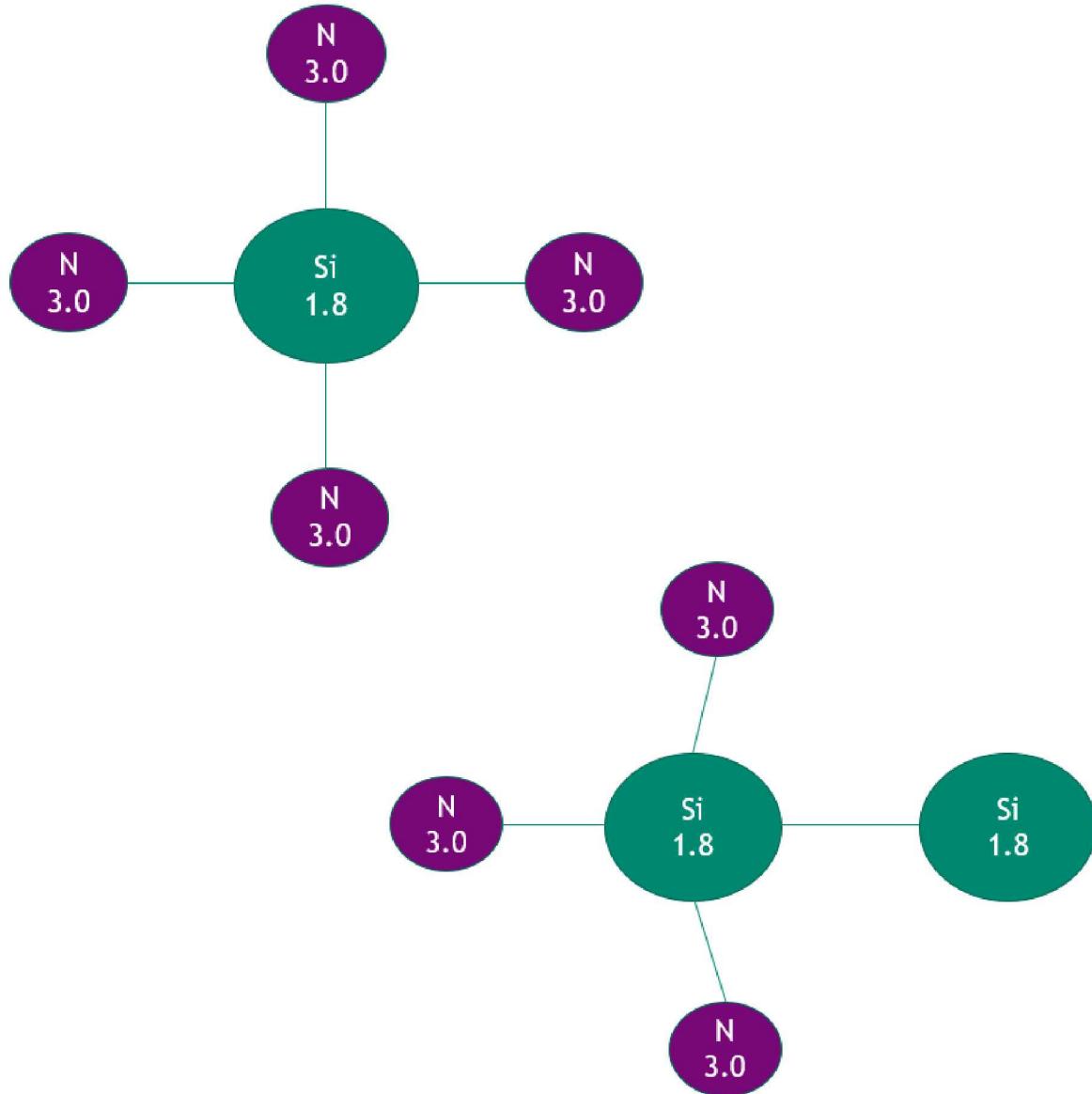


- We found that  $K$  is NOT a constant parameter across varying compositions
- This means that it cannot be used to solve for  $[N]$  for any  $SiNx$  film
- Verified statistically significant by linear regression significance testing
- Note that the values for  $K$  are thickness corrected values

Oscillator strength factors for films of varying composition and thickness

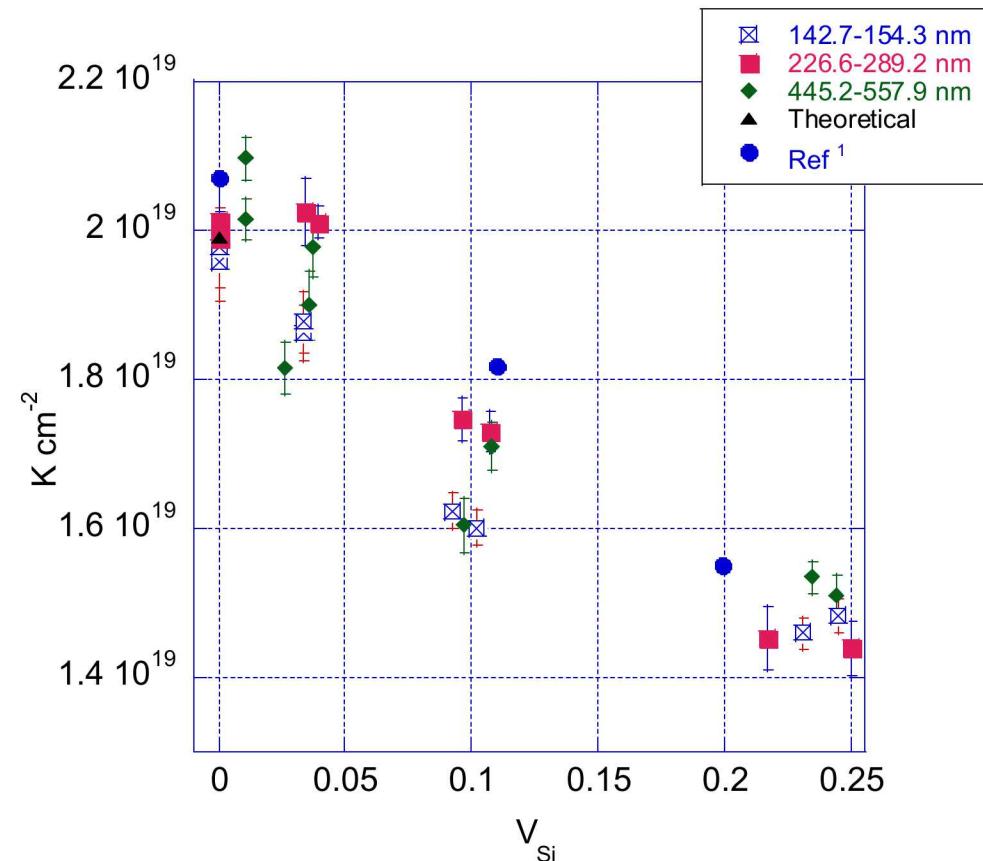
# K decreases as increasing $V_{Si}$ induces charge transfer

- The substitution of electronegative N atoms for a less  $e^-$  negative Si atom will modify the charge of other bonds to that atom
- The modified charge distribution induces structural changes to the basal tetrahedron
- Structural changes affect the bond strain which alters the effective dipole moment and modifies the IR absorption



# Trends observed with LPCVD films are similar to PECVD films

- Plasma enhanced chemical vapor deposition (PECVD)
- A value of  $2.07 \times 10^{19} \text{ cm}^{-2}$  is reported for a  $\sim 700 \text{ nm}$  thick stoichiometric  $\text{SiN}_x$  film deposited by PECVD
- This agrees with our results for films deposited by LPCVD
- Suggests that impurities introduced by PECVD do not effect the Si-N charge transfer
- K values appear to be independent of deposition method



# Theoretical K value was found by assuming $q=1.4e$

- A theoretical K value was found to be  $1.99 \times 10^{19} \text{ cm}^{-2}$  for a stoichiometric film
- $q$  typically ranges from  $.3e-3e$
- $q$  was found to be 1.4 times greater than 1e to accurately approximate K
- This value for  $q$  is within the expected range

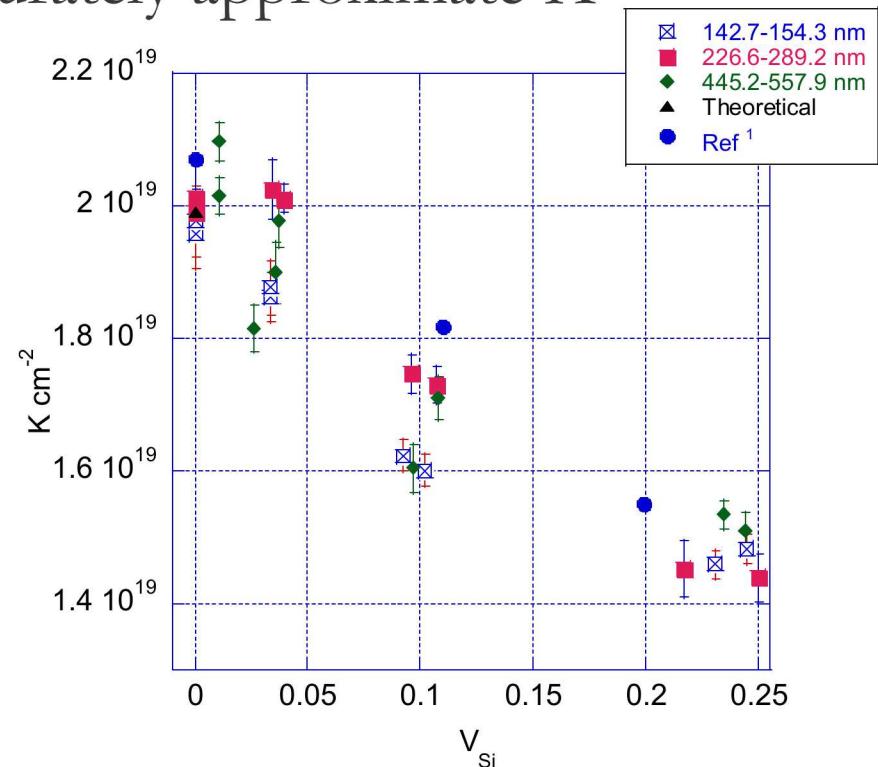
$$K \equiv \frac{c^2 n \omega_t \mu}{\pi q^2}$$

$$n=2.4$$

$$\omega_t=840 \text{ cm}^{-1}$$

$$\mu=1.56 \times 10^{19} \text{ kg}$$

$$q=6.72 \times 10^{-10} \text{ esu}$$



## Summary

- K values for films of varying composition and thickness were measured
- K decreases with increasing  $V_{Si}$
- Believed to be due to a charge transfer which affects the film strain
- Differences in strain alter the effective dipole moment which changes the IR properties
- Si-rich  $SiN_x$  films have a higher absorption probability
- K is a parameter which is ostensibly independent of deposition method



# Questions

