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# Combined computational and experimental study of zirconium tungstate

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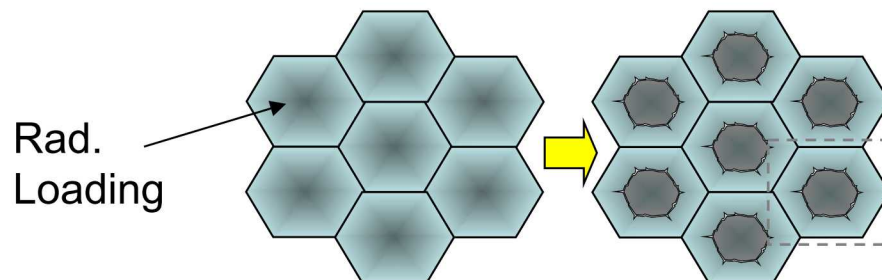
# Nuclear waste forms

- ❑ Radioactive materials:
  - 34 tons of weapons-grade Pu metal from the US weapons program
  - *Long-lived isotope of technetium*,  $^{99}\text{Tc}$  ( $t_{1/2} = 2.1 \times 10^5$  years,  $\beta^- = 0.294$  MeV; 6% of the fission yield for  $^{235}\text{U}$ )
- ❑ *Chemical and physical waste forms for immobilizing radionuclides* (e.g., U, Pu, Tc):
  - Pu-waste form to isolate the radionuclides for extended periods of time needed
  - No effective Tc waste form identified yet (e.g., *highly mobile* as soluble  $\text{TcO}_4^-$ , volatile at elevated temperatures such as vitrification)
- ❑ Ideal radioactive material waste forms:
  - Low solubility
  - Ability to isolate radionuclides from the environment for long periods of time
  - Resistant to radiation damage that can lead to expansion, crumbling, increased exposed surface area, thus more rapid dissolution
- ❑ Ceramic waste forms (e.g., Zr-W-O and/or Zr-W-P-O).

# Ceramic waste forms: $\text{ZrW}_2\text{O}_8$

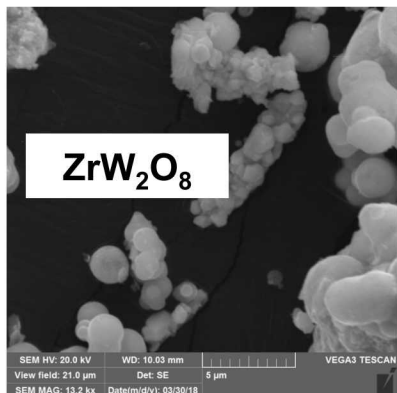
- ❑ Negative thermal expansion (NTE)
  - Contract upon heating
  - Naturally-occurring Zr minerals
  - Readily incorporate radionuclides (e.g., U, Tc, Pu) through substitution
- ❑ Radiation induced amorphization
- ❑ Zoned waste forms
  - No pressure exerted on the shell
  - No rupture of individual grains, mitigating the effect of amorphization on radionuclide release
  - Easy to maintain isolation of the radionuclide for long periods of time

**Good candidate for radiation resistant waste forms!**

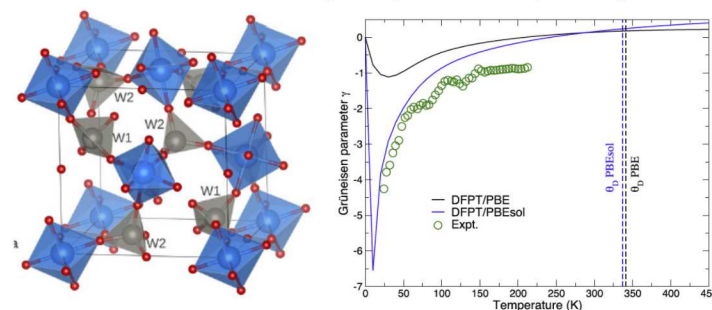


# Combined computational-experimental strategy

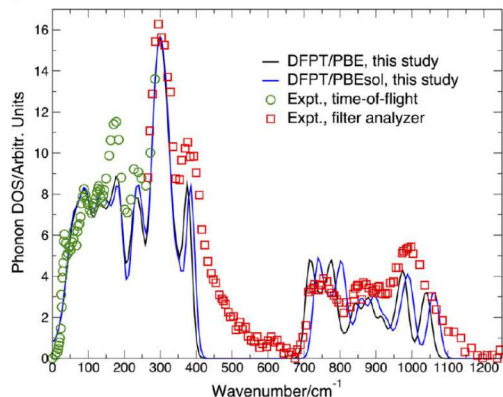
## Synthesis (cubic $\text{ZrW}_2\text{O}_8$ and $\text{Zr}_2\text{P}_2\text{WO}_{12}$ )



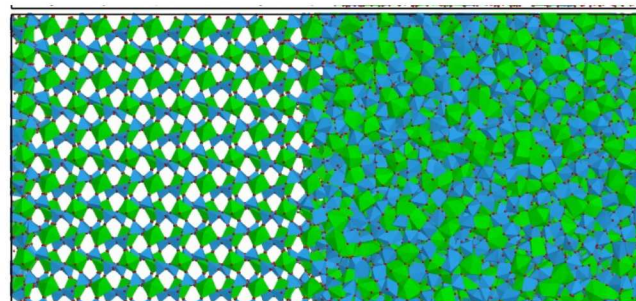
## DFT/DFPT calculations (structure, vibrational, thermodynamics, and thermo-mechanical properties, etc)



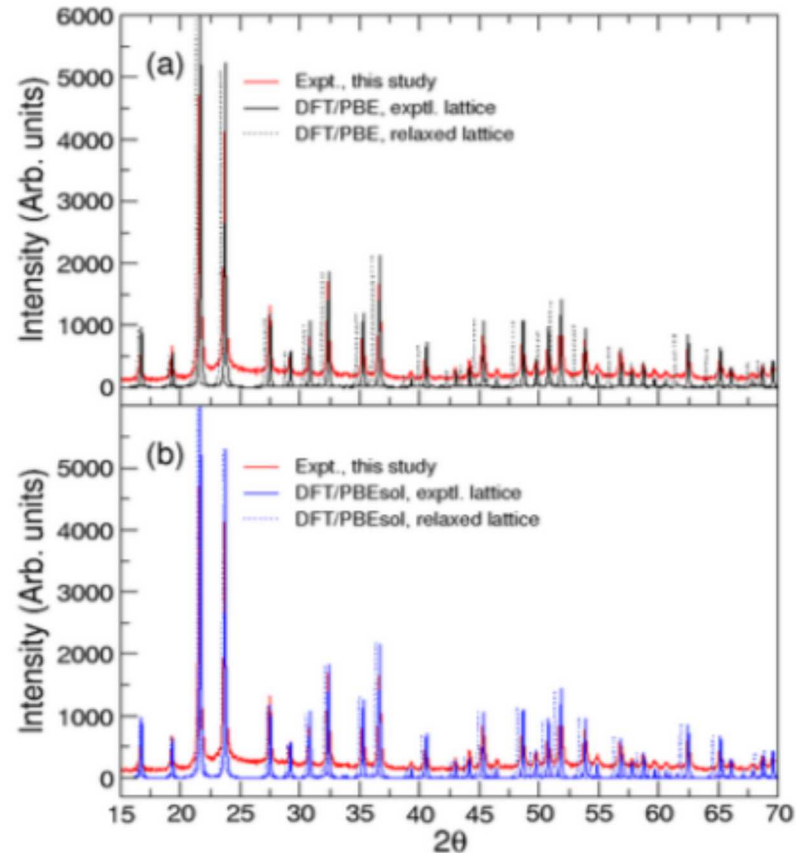
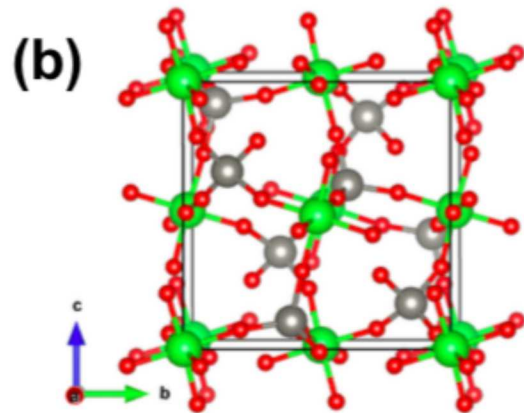
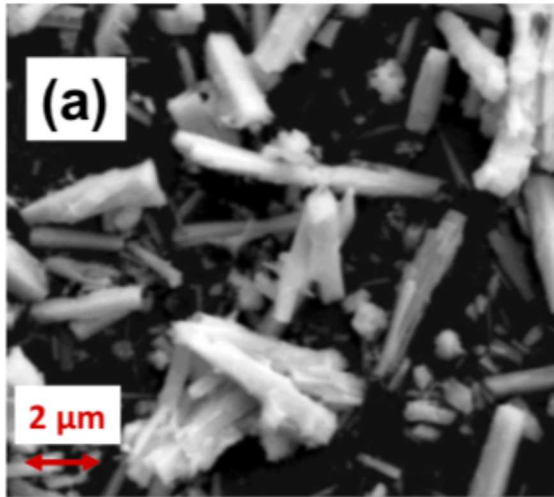
## Characterization (XRD, mid/far-IR, Raman)



## LAMMPS simulations (MD models for amorphization, crystal/amorphous material boundary)



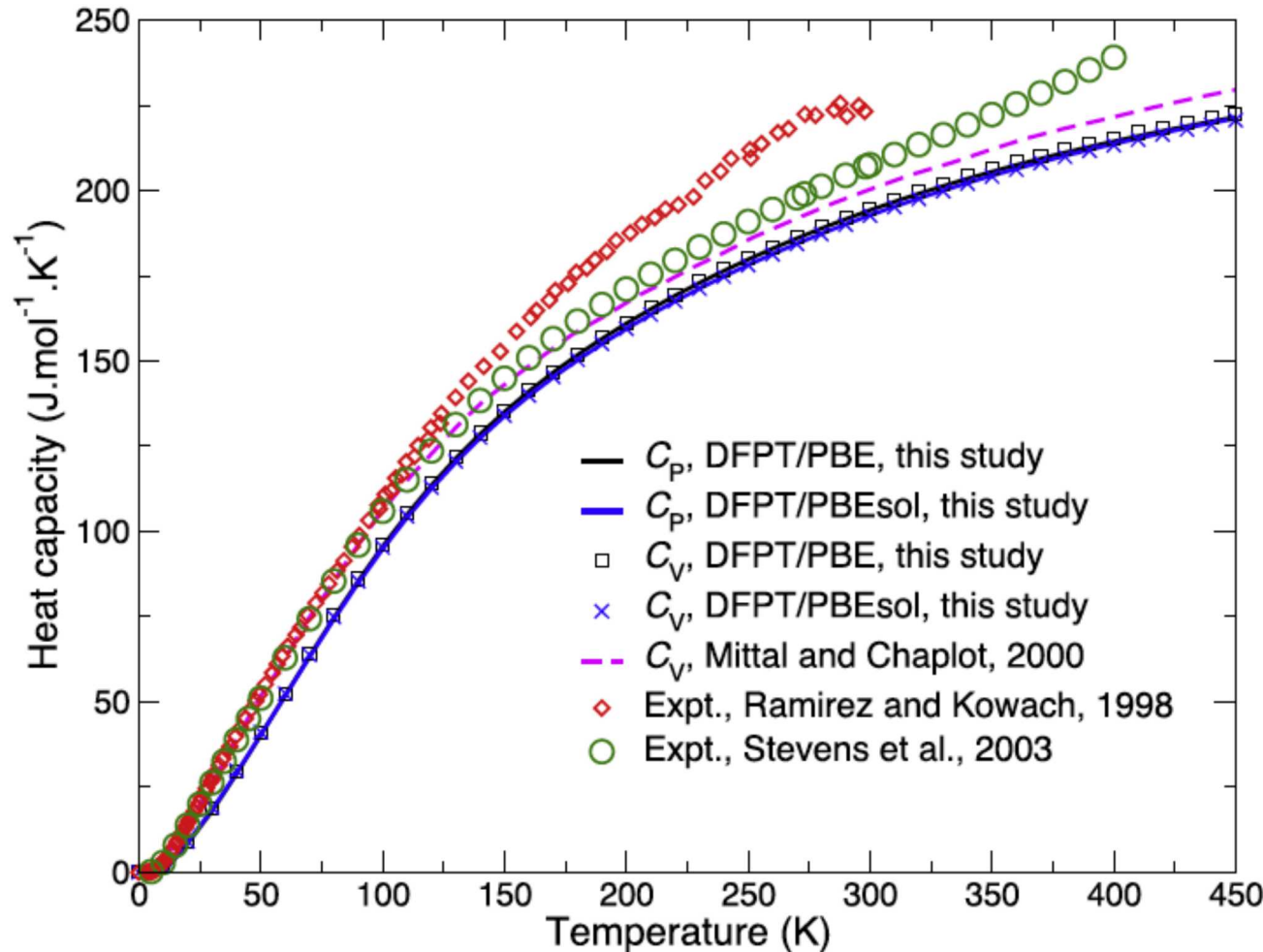
# Zr Tungstate



	DFT		Expt.	
	PBE	PEBsol	This work	Evans et al.
a (Å)	9.310	9.241	9.1493	9.1846

$\alpha$ -ZrW<sub>2</sub>O<sub>8</sub> (space group P2<sub>1</sub>3; IT No. 198; Z=4) with corner-sharing ZrO<sub>6</sub> octahedral and WO<sub>4</sub> tetrahedral coordination units [Weck et al., Chem. Phys. Lett. 698, 195 (2018)].

# Thermal properties of $\alpha\text{-ZrW}_2\text{O}_8$



$$C_p(T,P) = -T \frac{\partial^2 G(T,P)}{\partial T^2}$$

$$= T \left. \frac{\partial V(T,P)}{\partial T} \frac{\partial S(T,V)}{\partial V} \right|_{V=V(T,P)} + C_v[T,V(T,P)],$$

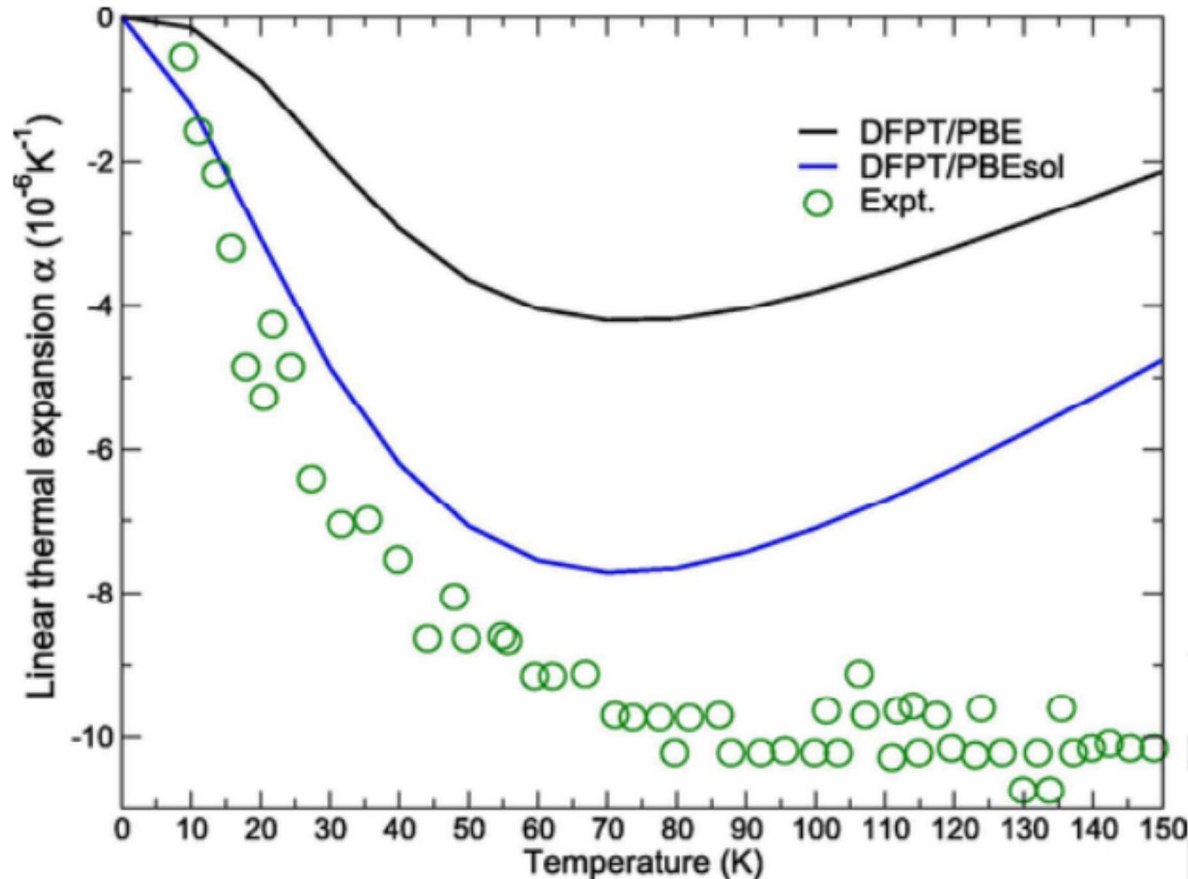
$$C_v = k_B \sum (\beta \hbar \omega)^2 \frac{e^{\beta \hbar \omega}}{[e^{\beta \hbar \omega} - 1]^2}$$

	$C_p^0$ (J/mol/K)
DFT	193.8 [1]
Expt	207.01[2]; 223.3 [3]

- [1] Weck et al., 2018.  
 [2] Stevens et al., 2003.  
 [3] Ramirez and Kowach, 1998.

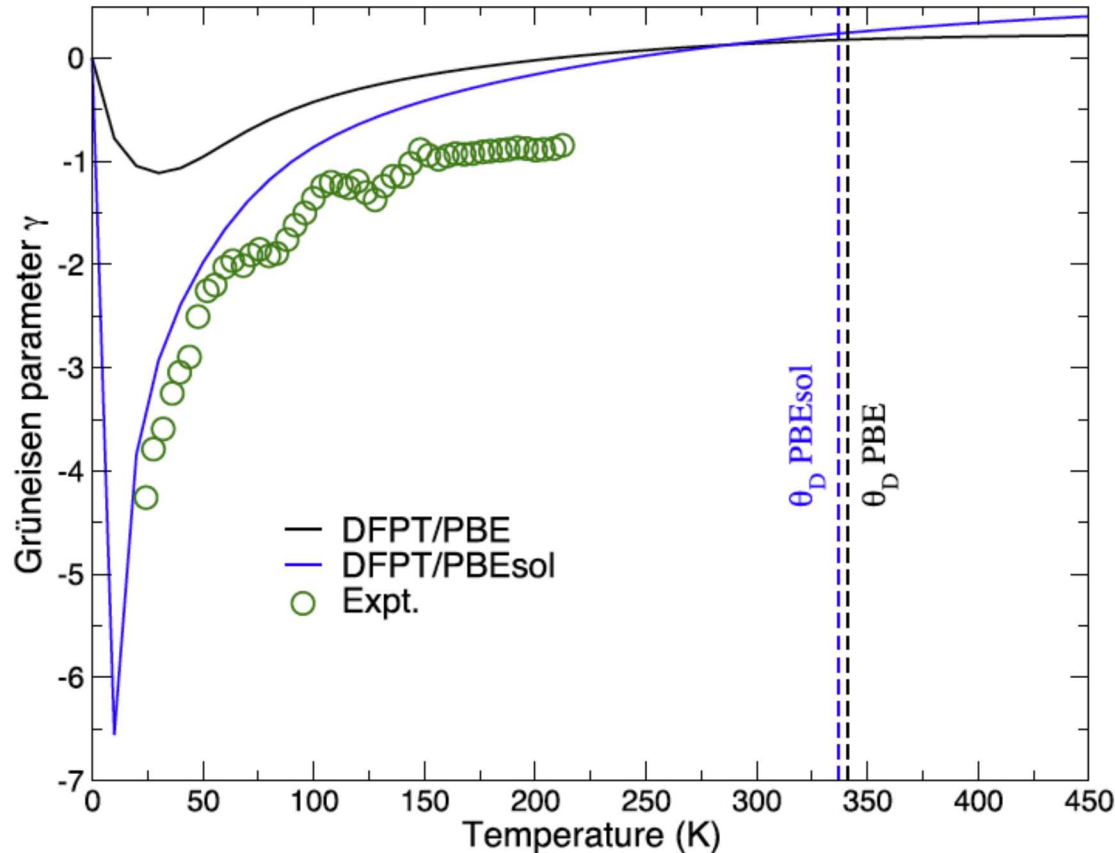
Thermal evolution of the molar isochoric ( $C_v$ ) and isobaric ( $C_p$ ) heat capacities of  $\alpha\text{-ZrW}_2\text{O}_8$  calculated at constant atmospheric pressure using DFT at the GGA/PBE and GGA/PBEsol levels within the quasi-harmonic approximation.

# Linear coefficient of thermal expansion (CTE) of $\alpha\text{-ZrW}_2\text{O}_8$



- The linear CTE computed using PBEsol is in very good agreement with the experimental data from the high-resolution neutron diffraction measurements of David et al. (1999) to  $\sim 60\text{K}$
- The PBEsol linear CTE reaches an extremum of at  $\sim 70\text{K}$
- Similar discrepancies were found above  $\sim 60\text{K}$  between these experimental data [6] and the finite-displacement B3LYP calculations of Gava et al. (2012)

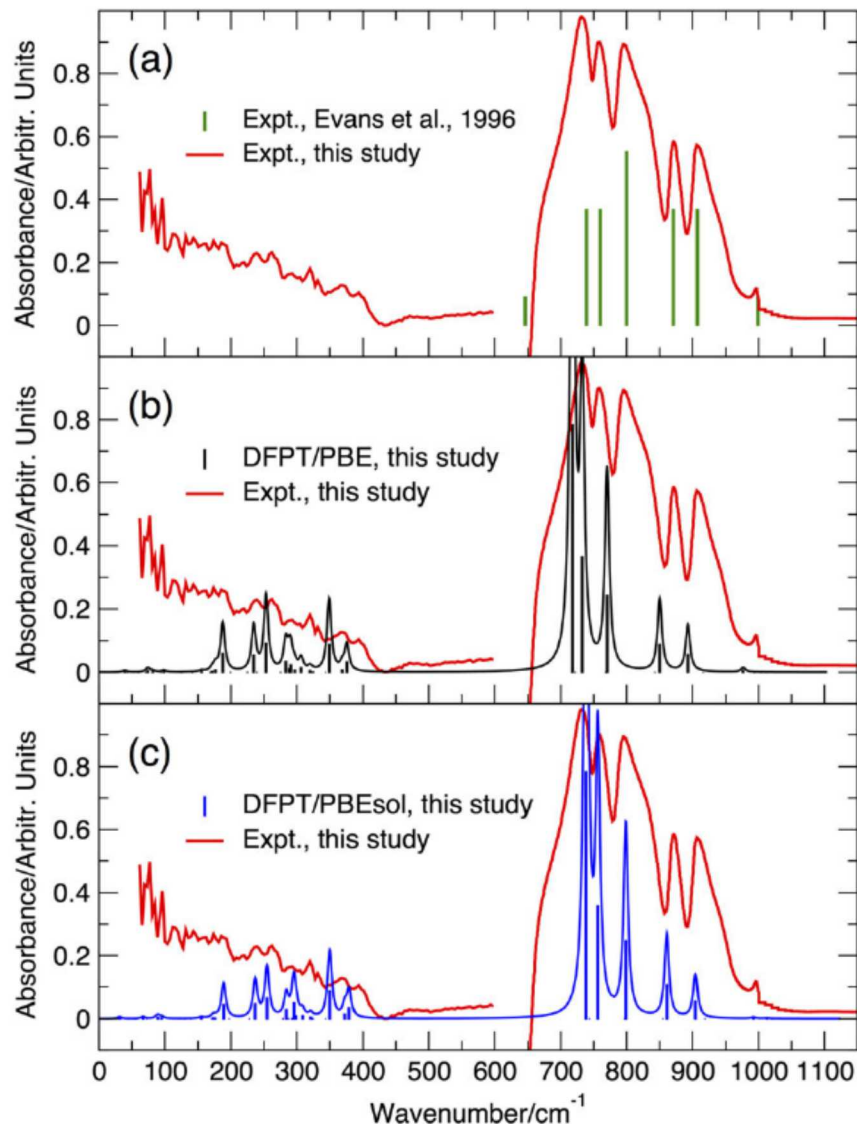
# Grüneisen parameter of $\alpha\text{-ZrW}_2\text{O}_8$



- PBEsol results are in excellent agreement with the experimental Grüneisen parameter by Ernst et al. (1998) to 60K.
- The main contributions to NTE at low temperature stem from the two lowest-frequency optical phonons, predicted at  $36\text{ cm}^{-1}$  ( $T_{\text{irrep}}$ ) and  $37\text{ cm}^{-1}$  ( $E_{\text{irrep}}$ ).
- These modes are characterized by variations of intrapolyhedral bond angles O-M-O (M = W, Zr), achieved through combined librational and translational motions of  $\text{WO}_4$  units.

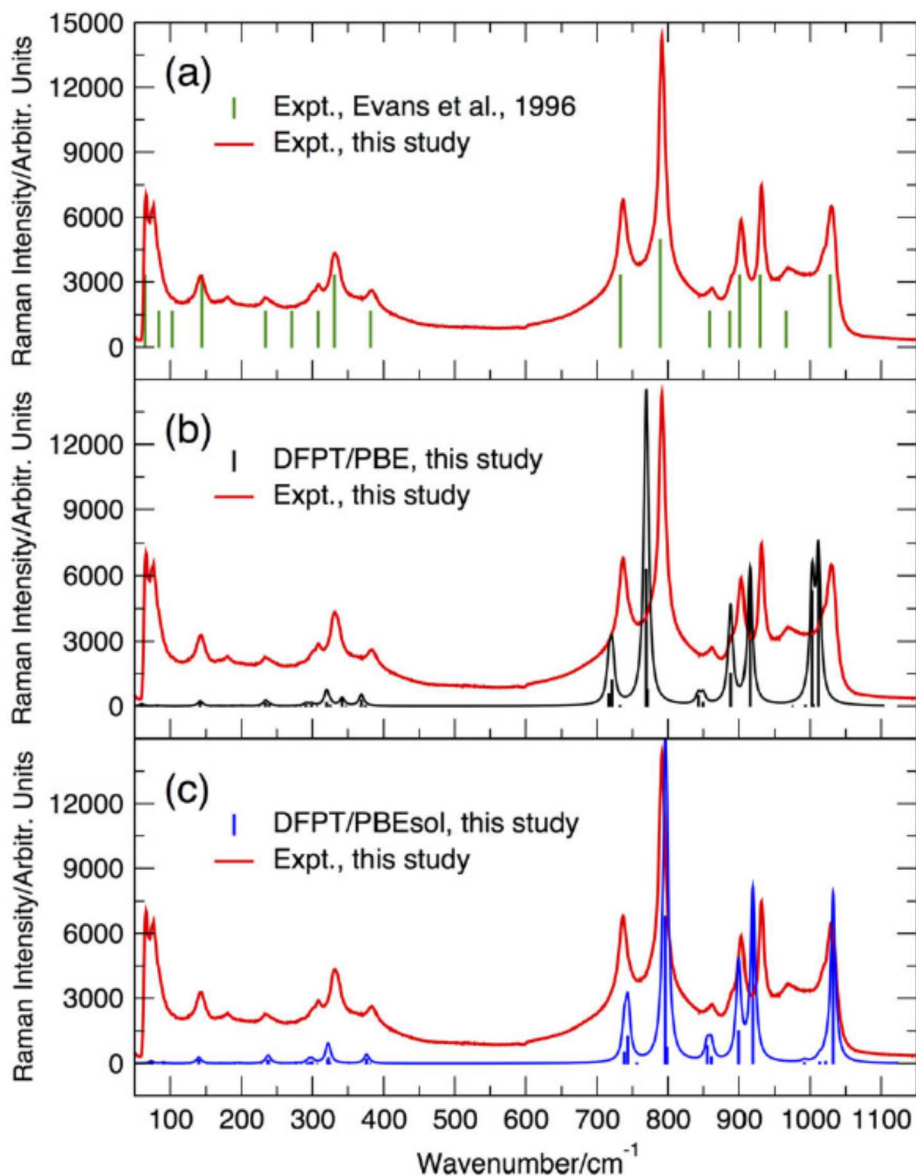
Thermal evolution of the Grüneisen parameter of  $\alpha\text{-ZrW}_2\text{O}_8$  calculated using DFT at the GGA/PBE and GGA/PBEsol levels within the quasi-harmonic approximation.

# IR spectrum of $\alpha$ -ZrW<sub>2</sub>O<sub>8</sub>



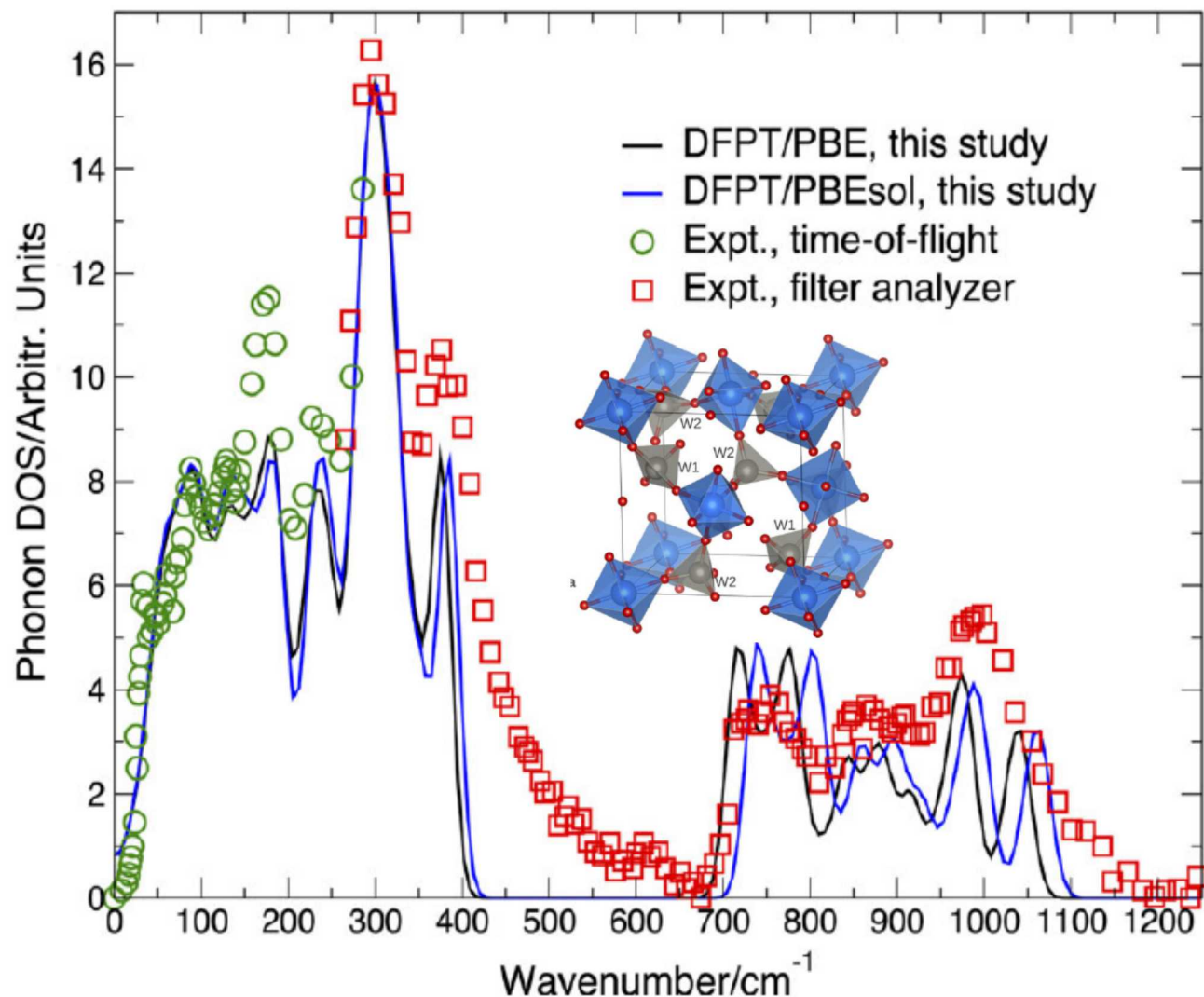
- Fourier transform infrared (FT-IR) spectra of  $\alpha$ -ZrW<sub>2</sub>O<sub>8</sub> measured in this study at T = 298 K (red line) and observed IR band centers reported by Evans et al. (green lines), along with infrared spectra simulated from DFPT at the GGA/PBE (black) and GGA/PBESol (blue) levels
- The IR spectrum simulated using PBESol is in close agreement with experiments
- The IR-active wavenumber observed by Evans et al. at 646 cm<sup>-1</sup> along with several weak lines in the range 600–400 cm<sup>-1</sup> were not predicted by DFPT calculations, nor observed in the present experiments

# Raman spectrum of $\alpha\text{-ZrW}_2\text{O}_8$



- Raman spectrum of  $\alpha\text{-ZrW}_2\text{O}_8$  measured in this study at  $T = 298\text{ K}$  (red) and observed and calculated Raman band centers reported by Evans et al. (green), along with Raman spectra simulated from DFPT-PBE (black) and DFPT-PBEsol (blue) levels
- No  $855\text{--}628\text{ cm}^{-1}$  peaks appear in the  $\alpha\text{-ZrW}_2\text{O}_8$  Raman bands predicted with DFPT or measured in this study or by Evans et al
- No stable  $\text{ZrO}_2$  or  $\text{WO}_3$  present in the sample
- Similar to PDOS and IR DFPT calculations, no Raman active modes are predicted to occur in the ranges  $381\text{--}739\text{ cm}^{-1}$  and  $374\text{--}717\text{ cm}^{-1}$  with PBEsol and PBE, respectively

# Phonon density of states (PDOS) of $\alpha$ -ZrW<sub>2</sub>O<sub>8</sub>



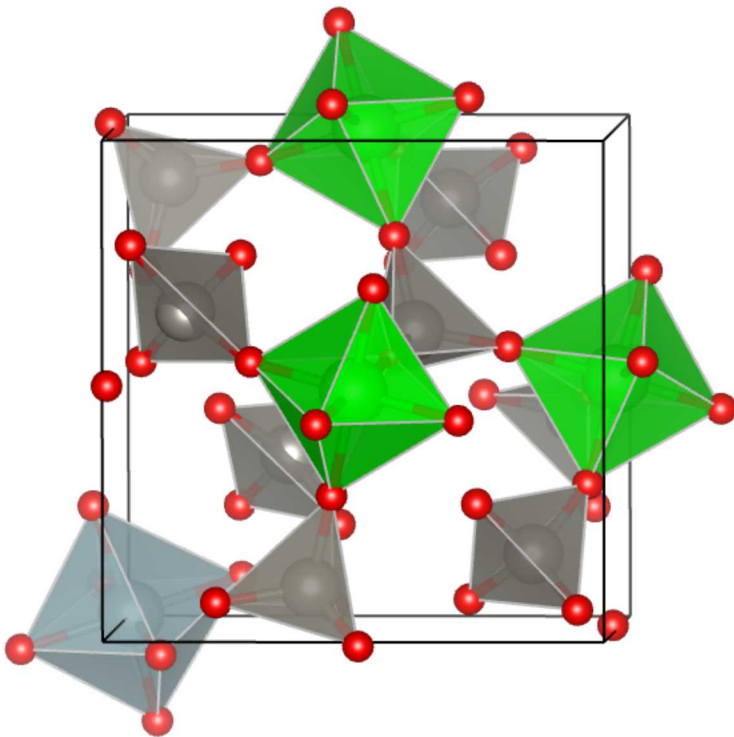
Phonon density-of-states (PDOS) spectra of  $\alpha$ -ZrW<sub>2</sub>O<sub>8</sub> simulated at T = 0 K using density functional perturbation theory (DFPT) at the GGA/PBE (black line) and GGA/PBEsol (blue line) levels. The generalized PDOS spectrum from inelastic neutron scattering data collected at T = 300 K using time-of-flight (green circles) and filter-analyzer (red squares) spectroscopy is shown for comparison

# Conclusion

- DFPT calculations were conducted at the GGA/PBE and GGA/PBESol levels of theory to systematically investigate the crystal structure, phonon, IR and Raman spectra, and vibrational properties of  $\alpha$ -ZrW<sub>2</sub>O<sub>8</sub>
- DFPT calculations were able to correctly reproduce the NTE behavior, elastic/mechanical properties, and IR/Raman spectra of ZrW<sub>2</sub>O<sub>8</sub>
- The calculated thermodynamic/thermo-mechanical properties were validated with measured data
- Calculations for Zr<sub>2</sub>P<sub>2</sub>WO<sub>12</sub> are being finalized and also appear to reproduce available experimental data

# Current/future research

## 1. U/Pu/Tc inclusion in $\alpha$ - $\text{ZrW}_2\text{O}_8$



## 2. DFT Modeling of $\text{Zr}_2\text{P}_2\text{WO}_{12}$

